


Cite this: *Digital Discovery*, 2026, 5, 123

# Extrapolating beyond C<sub>60</sub>: advancing prediction of fullerene isomers with FullereneNet

Bin Liu,<sup>†ab</sup> Jirui Jin<sup>†ab</sup> and Mingjie Liu <sup>\*ab</sup>

Fullerenes, carbon-based nanomaterials with sp<sup>2</sup>-hybridized carbon atoms arranged in polyhedral cages, exhibit diverse isomeric structures with promising applications in optoelectronics, solar cells, and medicine. However, the vast number of possible fullerene isomers complicates efficient property prediction. In this study, we introduce FullereneNet, a graph neural network-based model that predicts fundamental properties of fullerenes using topological features derived solely from unoptimized structures, eliminating the need for computationally expensive quantum chemistry optimizations. The model leverages topological representations based on the chemical environments of pentagonal and hexagonal rings, enabling efficient capture of local structural details. We show that this approach yields superior performance in predicting the C–C binding energy for a wide range of fullerene sizes, achieving mean absolute errors of 3 meV per atom for C<sub>60</sub>, 4 meV per atom for C<sub>70</sub>, and 6 meV per atom for C<sub>72</sub>–C<sub>100</sub>, surpassing the values of the state-of-the-art machine learning interatomic potential GAP-20. Additionally, the FullereneNet model accurately predicts 11 other properties, including the HOMO–LUMO gap and solvation free energy, demonstrating robustness and transferability across fullerene types. This work provides a computationally efficient framework for high-throughput screening of fullerene candidates and establishes a foundation for future data-driven studies in fullerene chemistry.

Received 31st May 2025  
Accepted 30th October 2025

DOI: 10.1039/d5dd00241a

rsc.li/digitaldiscovery

## 1 Introduction

Fullerenes are carbon-based nanomaterials composed of sp<sup>2</sup> hybridized carbons with the general formula C<sub>20+2n</sub> ( $n \geq 0$ ,  $n \neq 1$ ).<sup>1,2</sup> These carbon atoms form pentagonal and hexagonal rings on the spherical polyhedral cages. The different arrangements of these rings lead to numerous isomeric structures for each fullerene size, with the number of possible isomers increasing at a rate of O(N<sup>9</sup>) for N = 20 + 2n carbon atoms, resulting in a diverse family of fullerenes.<sup>3</sup> Thanks to their unique spherical cage structure and remarkable physicochemical properties, various fullerenes have been synthesized and applied in optoelectronics, solar cells, gas storage and separation, biology, and medicine.<sup>4–11</sup> For example, fullerenes—C<sub>60</sub>, C<sub>70</sub>, and C<sub>84</sub>—and their derivatives, which conform to the isolated pentagon rule (IPR) with 12 pentagons separated by hexagons to minimize steric strain and ensure high stability, are used as electron acceptors in organic solar cells.<sup>12,13</sup> Pristine non-IPR fullerenes, such as C<sub>20</sub> and C<sub>36</sub>, which are unstable and highly reactive due to their condensed pentagons and increased strain, have been successfully synthesized and isolated.<sup>14,15</sup> Additionally, non-IPR fullerenes with endohedral functionalization (encapsulation of

metal and/or nonmetal atoms, *e.g.*, Sc<sub>2</sub>@C<sub>66</sub> and Sc<sub>3</sub>N@C<sub>68</sub>) or exohedral functionalization (anchoring functional groups on the cage surface, *e.g.*, chlorinated C<sub>50</sub>Cl<sub>10</sub>, C<sub>60</sub>Cl<sub>12</sub>, and C<sub>60</sub>Cl<sub>8</sub>) exhibit unusual electronic, magnetic, and mechanical properties.<sup>16–19</sup> So far, only a small portion of fullerenes has been investigated for real-world applications through a laborious trial-and-error approach. To fully harness their capabilities in customizing electronic properties and functionalization for practical uses, it is crucial to accurately and efficiently predict the fundamental properties, including stability, electronic properties, and solubility, for high-throughput screening of potential fullerene candidates. However, the expansive fullerene family—with its range of sizes and numerous isomers for each size (*e.g.*, C<sub>84</sub> has 51 592 isomers)—not only poses significant challenges to mathematical enumeration and topological analysis<sup>20</sup> but also surpasses the computational capacity of current high-performance resources for quantum chemistry calculations.

To circumvent this computational bottleneck, early studies explored connections between the graph-theoretical properties of fullerenes and their physical characteristics. Schwerdtfeger *et al.*'s review<sup>21</sup> provides an extensive account of such topological indicators, demonstrating how graph-based indices derived from adjacency matrices can qualitatively predict properties, such as the HOMO–LUMO gap through approaches like Hückel theory, and even anticipate phenomena such as Jahn–Teller distortions. However, these traditional approaches rely heavily

<sup>a</sup>Department of Chemistry, University of Florida, Gainesville, FL 32611, USA. E-mail: mingjieliu@ufl.edu

<sup>b</sup>Quantum Theory Project, University of Florida, Gainesville, FL 32611, USA

<sup>†</sup> These authors contributed equally to this work.



on human-derived heuristics and predefined physical approximations, which constrain their predictive power to the limits of established chemical intuition.

Data-driven machine learning (ML) techniques have become powerful tools for predicting properties of fullerenes and their functionalized derivatives. For example, García-Risueño *et al.*<sup>22</sup> trained various conventional ML models to predict the highest occupied molecular orbital (HOMO), the lowest unoccupied molecular orbital (LUMO), and gap renormalization energies using a dataset of 163 fullerenes and their derivatives, ranging from C<sub>28</sub> to C<sub>180</sub>. They tested numerous variables as input representations, including density functional theory (DFT)-calculated electronic properties, geometric features, phonon features, bond length features, and bond order features. Liu *et al.*<sup>23</sup> trained a SchNet model-based neutral network potential (NNP), which uses molecular coordinates as input representations and a dataset of 120 k non-isomorphic C<sub>2n</sub>Cl<sub>m</sub> chloro-fullerene isomers generated from 500 pristine cages of C<sub>40</sub>, C<sub>50</sub>, C<sub>60</sub>, and C<sub>70</sub>. The NNP achieved a mean absolute error (MAE) of 0.37 eV for relative energy prediction with respect to DFT benchmarks and demonstrated excellent transferability to other exohedral functionalized fullerenes C<sub>2n</sub>X<sub>m</sub> (X = H, F, I, Cl, Br, OH, CF<sub>3</sub>, CH<sub>3</sub>) across different functional groups, number of addends, and cage sizes. However, previous studies have relied on DFT-calculated features or those derived from DFT-optimized geometry for ML training, which requires significant computational cost, making it unsuitable for high-throughput screening of fullerene candidates. Machine Learning Interatomic Potentials (MLIPs) have also been employed for fullerene systems. Aghajamali and Karton<sup>24</sup> applied the Gaussian Approximation Potential (GAP-20)<sup>25</sup> force field to investigate the isomerization energies and thermal stabilities of C<sub>40</sub> fullerenes, while Karasulu *et al.*<sup>26</sup> used GAP-20 to predict large carbon clusters and search for novel isomers, overcoming the computational limits of first-principles approaches. However, GAP-20 is limited to energy-potential surface prediction and cannot predict other fundamental properties which are crucial for real-world applications, such as molecular orbital levels, HOMO–LUMO gaps, and solubilities. More importantly, employing MLIPs requires geometry optimization to determine ground-state energies, resulting in additional computational costs.

Graph neural networks (GNNs) have gained considerable attention in chemistry due to their exceptional performance in molecular and materials science.<sup>27–30</sup> Their effectiveness stems from the ability to represent molecules and materials as graphs, where nodes correspond to atoms and edges to bonds. For example, TensorNet,<sup>31</sup> which utilizes Cartesian tensor representations, demonstrates state-of-the-art (SOTA) performance on the small organic molecule dataset QM9.<sup>32</sup> MatFormer,<sup>33</sup> incorporating a transformer architecture for learning periodic graph representations, is considered a leading GNN model for predicting crystal properties such as formation energy and band gap. However, these models require optimized structures as inputs, since precise structural information is essential for training, significantly increasing computational costs. Moreover, fullerenes, consisting solely of carbon, present unique

challenges due to identical node features, complicating the design of effective GNN models for this system. To the best of our knowledge, no existing GNN model can accurately and efficiently predict a wide range of fundamental properties for fullerenes of any size without relying on quantum chemistry-optimized structures as inputs.

In this work, we developed a GNN-based model to predict the fundamental properties of fullerenes using two sets of topological features based on the arrangement of pentagons and hexagons over the cage surface, as proposed in our previous study.<sup>34</sup> These topological features rely solely on carbon atom connectivity and can be efficiently extracted from unoptimized geometries, eliminating the need for quantum chemistry-based geometry optimization. Our results show that these features can effectively capture the intricate local structural environments of carbon atoms in fullerene, enabling excellent accuracy and extrapolation capability for predicting the C–C binding energy beyond C<sub>60</sub>. Notably, our GNN model trained with a C<sub>20</sub>–C<sub>58</sub> dataset achieved MAEs of 3, 4, and 6 meV/atom for test sets C<sub>60</sub>, C<sub>70</sub>, and C<sub>72</sub>–C<sub>100</sub>, respectively, surpassing the accuracy of the SOTA MLIP GAP-20. More profoundly, with the same topological features as structural representations, our retrained GNN models demonstrated high prediction accuracy for 11 other properties, including the HOMO, LUMO, gap, solvation free energies, and log *P*. Our study highlights the superior capability of topological features for predicting various fundamental properties of fullerenes with excellent accuracy and transferability. Our study lays a fundamental basis for future data-driven research in fullerene chemistry.

## 2 Methodology

### 2.1 Dataset construction

In this study, we constructed three datasets for model training and testing. The C<sub>20</sub>–C<sub>60</sub> dataset, consisting of 5770 structures, was adopted from our previous work.<sup>34</sup> This dataset provides a complete set of C<sub>20</sub>–C<sub>60</sub> structures, with properties such as the HOMO–LUMO gap, formation energy, and IP-EA computed using DFT (with the same methods applied in this work). The C<sub>70</sub> dataset, comprising 8149 isomers, was generated using the FULLERENE program (version 4.5) and subsequently optimized with a harmonic oscillator approximation-based force field,<sup>35</sup> which provides a good compromise between computational efficiency and accuracy in reproducing fullerene geometries. The C<sub>72</sub>–C<sub>100</sub> dataset (except C<sub>88</sub>) with 1171 IPR-conforming isomers was obtained from the Fullerene Library created by M. Yoshida.<sup>36</sup> Tables S1 and S2 list the number of possible isomers, C–C bonds, and hexagonal rings for each fullerene size in each set.

### 2.2 Computational details

We applied the Gaussian 16 package<sup>37</sup> for all the DFT calculations. We used the B3LYP hybrid functional<sup>38–40</sup> with D3 dispersion correction<sup>41</sup> and the 6-31G\* basis set for geometry optimization. The maximum force threshold was 0.02 eV Å<sup>−1</sup>. Then, we conducted a single-point calculation in the singlet state with the B3LYP functional and 6-311G\* basis set to



compute energies and electronic properties of the isomers. The solvation energies were calculated using the Solvation Model based on Density (SMD).<sup>42</sup> The dielectric constants for water and ODCB solvents were set at 78 and 10, respectively.

Geometry optimization with the GAP-20 potential was performed using the Large-scale Atomic/Molecular Massively Parallel Simulator (LAMMPS) package,<sup>43</sup> compiled with the Quantum mechanics and Interatomic Potentials (QUIP) library<sup>44,45</sup> including GAP routines. Geometry relaxation and energy minimization were conducted with the conjugate gradient method. The convergence criteria were set to  $10^{-4}$  eV for the total energy and  $10^{-6}$  eV Å<sup>-1</sup> for the atomic forces.

### 2.3 Feature space construction

We used atom and bond representations from our previous work<sup>34</sup> as node and edge features for GNN model training. In fullerene molecules, each carbon atom is fused to three adjacent polygonal rings, and each C–C bond is surrounded by four polygonal rings, either pentagons or hexagons. Atom features categorize carbon atoms into four types  $\{v_i | i = 0, 1, 2, 3\}$ , based on the number of adjacent pentagons (ranging from 0 to 3). The node feature is one-hot encoded with four digits to represent the local geometry of each carbon atom. Similarly, bond features categorize C–C bonds into nine types  $\{e_i | i = 0, \dots, 8\}$ , based on the number and arrangement of the surrounding polygonal rings. The edge feature is one-hot encoded with nine digits to indicate the local geometry of each C–C bond. The detailed atom and bond features are shown in Fig. 1.

Beyond the canonical fullerenes studied in this work, our methodology may also be applied to non-canonical fullerenes that include other polygons such as heptagons and octagons. While the present study focused exclusively on canonical fullerenes, we propose an extension of the current methodology

to address non-canonical cases (see Section S1), which will be carefully investigated in future work.

### 2.4 GNN model architecture

Here, we present our model named FullereneNet, inspired by the transformer-based architecture utilized in MatFormer<sup>33</sup> (model details are in Section S2). This model is designed to eliminate the need for quantum mechanical optimization of fullerene structures, as illustrated in Fig. 2. We define the molecular graph as  $G = (V, E)$ , where  $V$  and  $E$  represent the set of atoms and bonds in a molecule, respectively. Let  $v_i \in V$  denote the node feature of atom  $i$ , and  $e_{ij} \in E$  represent the edge feature between atoms  $i$  and its neighbor atom  $j$ . The overall model architecture consists of an embedding layer, multiple convolutional layers, and a readout function. The embedding layer projects the atomic feature  $v_i$  to a higher-dimensional vector, denoted as  $h_i$ . The convolutional layers are the core of the GNN model, where message passing occurs.<sup>46,47</sup> This process is essential for capturing interactions between atoms. The readout function aggregates the updated atomic representations into a molecular-level feature, which is then used for property prediction *via* a pooling function.

Our proposed message passing scheme is composed of three steps: attention score computation, attention coefficient normalization, and message computation with node information updating. In the first step, query  $Q_i$ , key  $K_j$ , and value  $V_j$  are calculated following the regular attention mechanism.<sup>48</sup> These vectors serve to evaluate how relevant each neighboring node is to the node being updated. By comparing the query of a node with the keys of its neighbors, the model calculates attention scores that determine the influence each neighbor should have. Specifically,

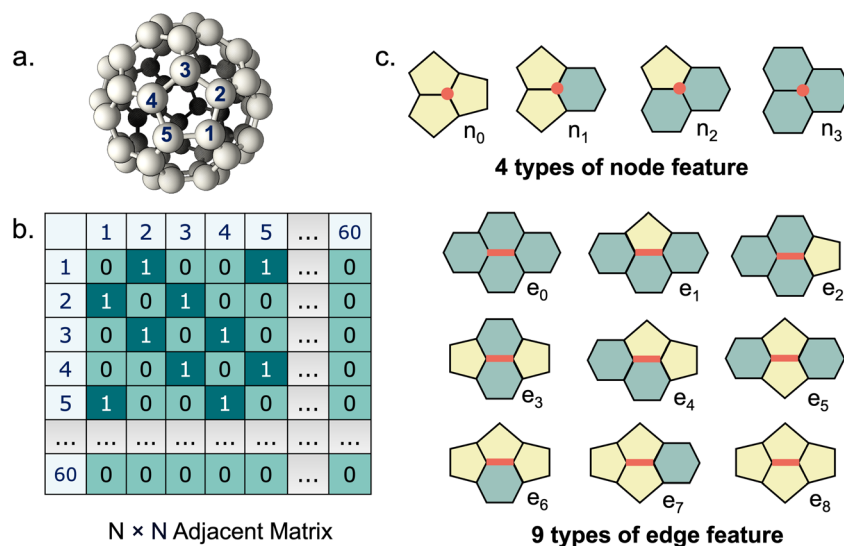


Fig. 1 Feature construction. (a) Taking C60 as an example, each atom in the fullerene is assigned a unique sequence number. (b) An adjacency matrix is generated, where entries 1 indicate a connection between atoms and entries 0 indicate no connection. (c) Topological node and edge features are derived from the adjacency matrix.



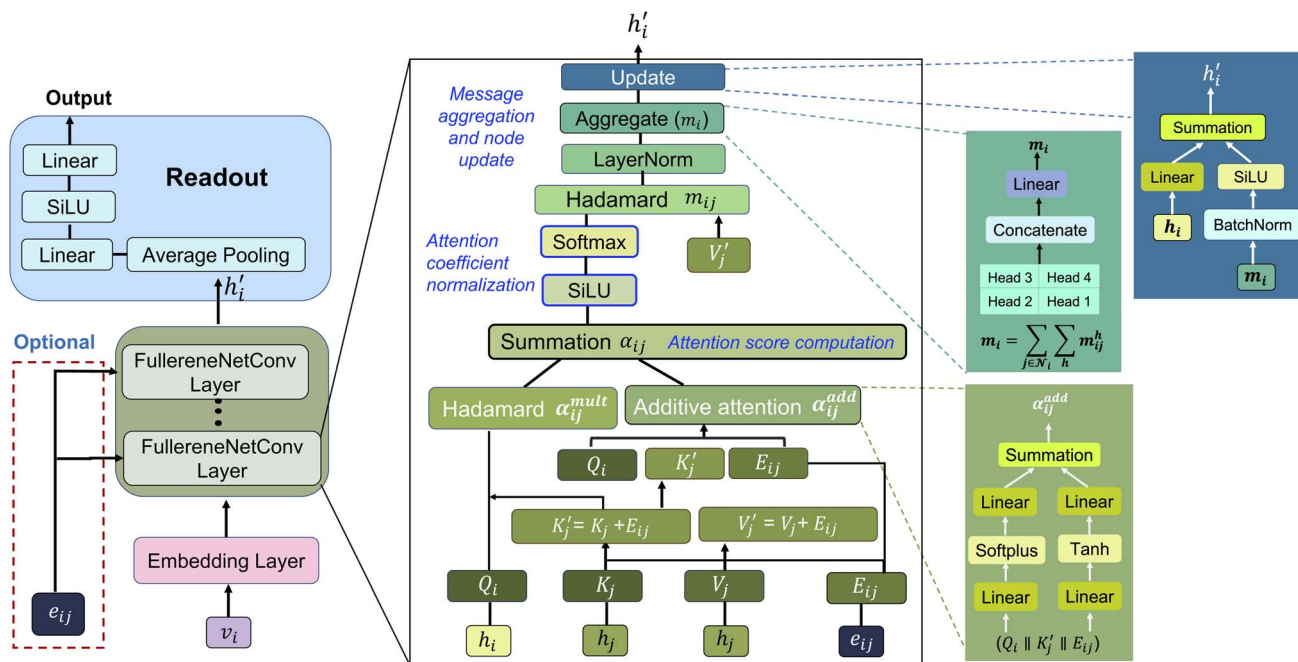


Fig. 2 The detailed architecture of FullereneNet.

$$Q_i = \text{LN}_{Q}(h_i), K_j = \text{LN}_{K}(h_j), V_j = \text{LN}_{V}(h_j), \text{ and } E_{ij} = \text{LN}_{E}(e_{ij}) \quad (1)$$

where  $\text{LN}_Q$ ,  $\text{LN}_K$ ,  $\text{LN}_V$ , and  $\text{LN}_E$  denote the linear transformations to compute query, key, value, and edge embedding. Then,  $K'_j$  and  $V'_j$  are obtained by summing  $K_j$  or  $V_j$  with  $E_{ij}$ , respectively. Next, we compute the attention score using both additive and multiplicative components. The additive attention computes attention scores by processing the concatenation of  $Q_i$ ,  $K'_j$ , and  $E_{ij}$  through two separate multi-layer perceptrons (MLPs) with different activation functions:

$$\alpha_{ij}^{\text{add } 1} = \text{MLP}_{\text{add } 1}(Q_i \| K'_j \| E_{ij}), \quad (2)$$

$$\alpha_{ij}^{\text{add } 2} = \text{MLP}_{\text{add } 2}(Q_i \| K'_j \| E_{ij}), \quad (3)$$

$$\alpha_{ij}^{\text{add}} = \alpha_{ij}^{\text{add } 1} + \alpha_{ij}^{\text{add } 2}, \quad (4)$$

where  $\|$  denotes concatenation,  $\text{MLP}_{\text{add}}$  consists of linear layers with an activation function (e.g., tanh, softplus, etc.), and  $\alpha_{ij}^{\text{add}}$  is the additive attention score. Another part is multiplicative attention, which is computed by using the scaled Hadamard product:

$$\alpha_{ij}^{\text{mult}} = \frac{Q_i \odot K'_j}{\sqrt{d}} \quad (5)$$

where  $\odot$  represents the Hadamard product and  $d$  is the dimensionality of the key vectors. The overall attention score  $\alpha_{ij}$  is obtained by summing the  $\alpha_{ij}^{\text{add}}$  and  $\alpha_{ij}^{\text{mult}}$  components followed by an activation function.

After obtaining the attention score  $\alpha_{ij}$ , we then move to the second part. A non-linear activation function, such as SiLU, is first applied to the combined attention scores  $\alpha_{ij}$ . To ensure that the attention coefficients are properly normalized, a softmax function is applied over the neighbors of atom  $i$ :

$$\alpha_{ij} = \text{softmax}(\alpha_{ij}) = \frac{\exp(\alpha_{ij})}{\sum_{k \in \mathcal{N}(i)} \exp(\alpha_{ik})} \quad (6)$$

where  $\mathcal{N}(i)$  denotes the set of neighbor nodes of atom  $i$ .

Finally, in the third part, messages from neighbors are aggregated and updated to the original atom representations. The messages from neighbors are first computed by weighting the value vectors  $V'_j$  with the attention coefficients  $\alpha_{ij}$  using the Hadamard product:

$$m_{ij} = \alpha_{ij} \odot V'_j \quad (7)$$

The processed messages are then passed through a layer normalization function. Subsequently, the messages from each node and attention head are aggregated using a summation method:

$$m_i = \sum_{j \in \mathcal{N}(i)} \sum_h m_{ij}^h \quad (8)$$

The last step is the node update with a residual connection:

$$h'_i = \text{LN}_{\text{Lin}}(h_i) + \sigma(\text{BN}(m_i)) \quad (9)$$

where  $\text{LN}_{\text{Lin}}$  denotes linear transformations,  $\sigma$  is an activation function, and BN indicates batch normalization. The updated node representation is obtained by adding the original node features with the message  $m_i$ .



MatFormer introduces an effective transformer variant tailored for crystal graph learning. However, our proposed model, FullereneNet, diverges from MatFormer in terms of feature construction and message passing schemes. While MatFormer leverages elemental diversity to extract a broad range of atomic features, it struggles with fullerenes, which consist solely of carbon and exhibit uniform atomic properties. To overcome this limitation, we design topological atomic features that distinguish pentagon and hexagon arrangements around carbon atoms, effectively capturing structural variations and enhancing representation. Additionally, MatFormer takes a DFT-optimized structure to derive bond distances and angles, whereas FullereneNet, with its bespoke feature design, bypasses the need for geometry optimization *via ab initio* methods as input, significantly reducing computational costs. Regarding message passing, MatFormer combines multiple information streams through a complex triple concatenation process. In comparison, FullereneNet employs a streamlined method that separately processes atomic interactions through two complementary pathways—one capturing linear relationships and another capturing non-linear relationships between connected atoms. This dual-pathway design eliminates redundant computational steps while better preserving the essential topological information that determines fullerene stability. Given the uniform node degree (*i.e.*, three) in fullerenes, FullereneNet does not encounter the challenge to distinguish nodes with different degrees as MatFormer.<sup>33</sup> Instead, it benefits from applying softmax normalization, which enhances training stability and model focus. The hyperparameter search range for FullereneNet is summarized in Table S3.

## 2.5 Model training strategy for extrapolation

In this study, we aim to develop a GNN model with strong extrapolation capabilities for predicting the stability of larger

fullerenes. To achieve this, we trained and validated our model on a dataset of small fullerenes ranging from  $C_{20}$  to  $C_{58}$ , including 3958 distinct isomers. Three larger fullerene groups were utilized as test sets: the complete  $C_{60}$  dataset containing 1812 isomers, a  $C_{70}$  dataset consisting of 100 randomly selected non-IPR isomers from a total of 8149 entries, and a  $C_{72}$ – $C_{100}$  dataset comprising 1171 IPR-conforming isomers. The random selection of  $C_{70}$  isomers ensures unbiased sampling across the energy landscape while avoiding the prohibitive computational cost of evaluating all 8149 isomers, while the curated  $C_{72}$ – $C_{100}$  dataset provides validation against established structural benchmarks. Moreover, the inclusion of both IPR and non-IPR fullerenes in the test sets allows us to demonstrate the model's reliable extrapolation capability.

Previous studies<sup>49,50</sup> have shown that different data-splitting strategies can enhance a model's extrapolation ability. To explore their impact on improving the robustness and effectiveness of extrapolation, we applied four distinct strategies to partition the training and validation sets (Fig. 3).

**2.5.1 Leave-One-Group-Out (LOGO).** Following the approach described by Zhao *et al.*,<sup>50</sup> the dataset was partitioned based on the fullerene cage size (*i.e.*, total number of carbon atoms). For instance,  $C_{58}$  was used as the validation set, while the fullerenes ranging from  $C_{20}$  to  $C_{56}$  constituted the training set. By rotating the validation set across fullerene sizes from  $C_{20}$  to  $C_{58}$ , five GNN models were generated. Notably,  $C_{20}$ – $C_{50}$  fullerenes were grouped as a single validation set since their isomer counts are relatively small, and their combined total is comparable to the number of isomers found in each of the larger cages ( $C_{52}$ ,  $C_{54}$ ,  $C_{56}$ , and  $C_{58}$ ).

**2.5.2 Leave-One-Cluster-Out (LOCO).** In contrast to LOGO, where the dataset is categorized by fullerene cage size, the LOCO method proposed by Meredig *et al.*<sup>49</sup> employs the k-means clustering algorithm<sup>51</sup> to cluster the data into five distinct groups. Each cluster was sequentially used as the

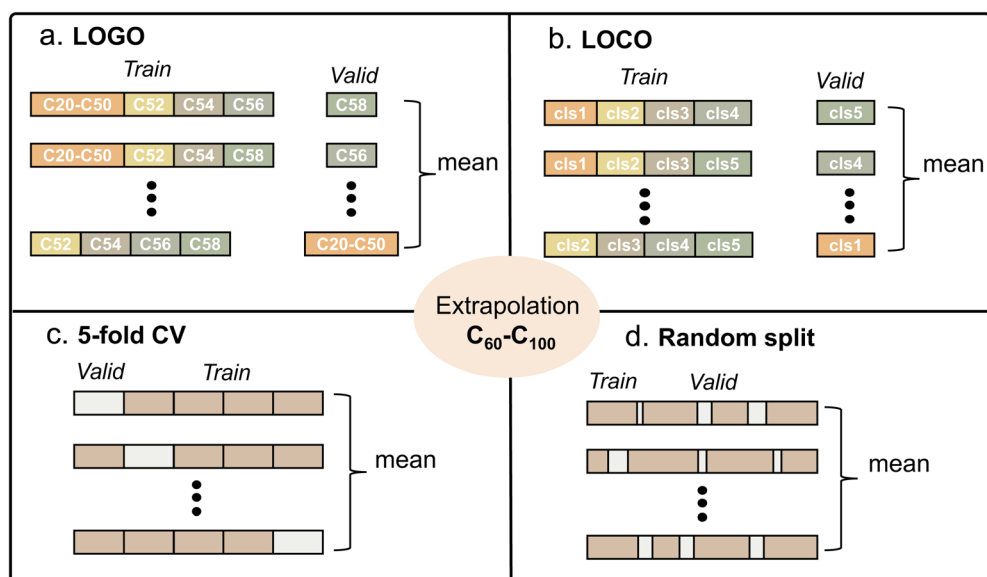


Fig. 3 Four different training strategies for extrapolation, including Leave-One-Group-Out (LOGO), Leave-One-Cluster-Out (LOCO), Five-fold Cross-Validation (5-fold CV), and random split, respectively.



validation set, while the remaining clusters were used for training, producing five GNN models.

**2.5.3 Five-fold cross-validation (5-fold CV).** The  $C_{20}$ – $C_{58}$  dataset was divided into training and validation sets using 5-fold cross-validation, resulting in the training of five GNN models.

**2.5.4 Random split.** The  $C_{20}$ – $C_{58}$  dataset was randomly partitioned into training and validation sets with a 4 : 1 ratio. Five GNN models were trained, each using a different random seed.

The extrapolation performance of each strategy was evaluated by averaging the prediction outcomes of the corresponding five GNN models across the three test sets: the  $C_{60}$ ,  $C_{70}$ , and  $C_{72}$ – $C_{100}$  datasets.

## 3 Results and discussion

### 3.1 Stability prediction

We evaluated the extrapolation performance of FullereneNet trained using the four strategies discussed in Section 2.5. The performance metrics for predicting the binding energies of fullerene in three test sets— $C_{60}$ ,  $C_{70}$ , and  $C_{72}$ – $C_{100}$ —are summarized in Table 1. For each strategy, the final prediction value was obtained by averaging the prediction values of the five corresponding GNN models on the test set. As shown in Table 1, all FullereneNet models trained on the  $C_{20}$ – $C_{58}$  dataset, irrespective of the training strategies employed, consistently demonstrate high accuracy in predicting binding energies of the  $C_{60}$ ,  $C_{70}$ , and  $C_{72}$ – $C_{100}$  datasets, as indicated by MAE and root mean squared error (RMSE) values. Notably, these models achieve small MAE values ranging from 3 meV per atom to 7 meV per atom.

The strong extrapolation capabilities of FullereneNet highlight the effectiveness of atom and bond features as input representations for predicting fullerene's stability. Based on our results, the four different data split strategies did not exhibit a statistically significant difference in their predictive performance for large-sized fullerenes. Furthermore, the prediction error for the  $C_{72}$ – $C_{100}$  test set was consistently larger than that

for the other two test sets, regardless of the training strategy employed. For instance, using the random split method, the FullereneNet model achieved a coefficient of determination ( $R^2$ ) of 0.563 and an MAE of 5 meV per atom for  $C_{72}$ – $C_{100}$ , compared to 0.989 and 3 meV per atom for  $C_{60}$  and 0.974 and 4 meV per atom for  $C_{70}$ . This slightly lower accuracy for the  $C_{72}$ – $C_{100}$  test set can be attributed to the fact that the binding energy distribution of the  $C_{72}$ – $C_{100}$  set is markedly different from that of the training set ( $C_{20}$ – $C_{58}$ ), as illustrated in Fig. S1.

As detailed in Section 2, FullereneNet incorporates node and edge features extracted from unoptimized fullerene structures as input. In contrast, MatFormer, the predecessor of FullereneNet, relies on atomic attributes such as atomic volume, valence electron count, and bond distances calculated from optimized Cartesian coordinates. However, since fullerene consists exclusively of carbon atoms, these conventional atomic descriptors become uniform across the structure, limiting their discriminative power for property prediction. Herein, we evaluated the performance of MatFormer, trained on the bond distance obtained from atomic coordinates of unoptimized fullerene structures, against FullereneNet in predicting the binding energies of fullerenes ranging from  $C_{60}$  to  $C_{100}$ . As shown in Fig. S2, MatFormer performs poorly with unoptimized structures, whereas optimizing fullerene structures significantly improves the prediction accuracy. For example, in the  $C_{70}$  dataset, utilizing optimized structures to train MatFormer increases the  $R^2$  value from 0.542 to 0.977 and reduces the MAE from 19 meV per atom to 4 meV per atom, as illustrated in Fig. S3.

The state-of-the-art MLIP, GAP-20, has been developed to accurately and efficiently predict isomerization energies, assess thermal stability, and identify new carbon clusters and fullerene isomers.<sup>26</sup> We evaluated the performance of GAP-20 on DTF optimized structures by predicting the relative binding energies for the  $C_{60}$ ,  $C_{70}$ , and  $C_{72}$ – $C_{100}$  datasets, using the binding energy of  $C_{60}$ -isomer-1 as the reference (Fig. S4). We emphasize that GAP-20 exhibits excellent accuracy in geometry optimization when benchmarked against the DFT method. For the  $C_{60}$  dataset, we confirmed strong correlations between relative binding energies calculated using GAP-20 with both DFT-optimized and GAP-20-optimized structures, achieving an impressive  $R^2$  value of 0.97 and a low MAE of 5 meV per atom (Fig. S5), consistent with previous studies.<sup>52</sup>

Fig. 4 presents a comprehensive comparison of binding energy prediction performance among FullereneNet (using unoptimized structures), MatFormer, and GAP-20 (both using optimized structures). Our results demonstrate that FullereneNet consistently outperforms the other two models across all three datasets. MatFormer exhibits poor performance on the  $C_{72}$ – $C_{100}$  test sets, achieving an MAE of 0.020 eV per atom. Similarly, GAP-20 shows suboptimal performance on the  $C_{60}$  test set, with an MAE of 0.016 eV per atom. Detailed MAE and  $R^2$  values for all methods are summarized in Table S4.

Besides the accuracy, one significant benefit of our model is that it can avoid the huge computational cost associated with computing large-size fullerenes. Take one  $C_{720}$  isomer<sup>36</sup> as an example, which represents a computationally challenging system

**Table 1** Performance metrics ( $R^2$ , MAE, and RMSE) across fullerene datasets using four different validation methods, including Leave-One-Group-Out (LOGO), Leave-One-Cluster-Out (LOCO), Five-fold Cross-Validation (5-fold CV), and random split, respectively

Dataset	Method	$R^2$	MAE	RMSE
$C_{60}$	LOGO	0.988	0.003	0.004
	LOCO	0.988	0.003	0.004
	5-Fold CV	0.989	0.003	0.004
	Random split	<b>0.989</b>	<b>0.003</b>	<b>0.003</b>
$C_{70}$ (non-IPR)	LOGO	0.969	0.004	0.005
	LOCO	0.972	0.004	0.005
	5-Fold CV	0.973	0.004	0.005
	Random split	<b>0.974</b>	<b>0.004</b>	<b>0.005</b>
$C_{72}$ – $C_{100}$ (IPR)	LOGO	0.304	0.007	0.009
	LOCO	0.431	0.006	0.008
	5-Fold CV	0.391	0.006	0.008
	Random split	<b>0.563</b>	<b>0.005</b>	<b>0.007</b>



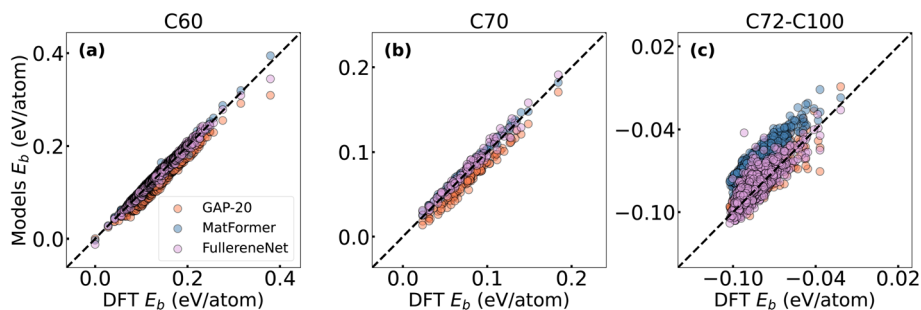


Fig. 4 Parity plots comparing the prediction results of FullereneNet (unoptimized structures), MatFormer, and GAP-20 (optimized structures) against DFT values, using  $C_{60}$ -isomer-1 as the reference. (a) Performance of the three models on the  $C_{60}$  test set. (b) Performance on the  $C_{70}$  test set. (c) Performance on the  $C_{72}$ – $C_{100}$  test set.

due to its substantial size. To quantify the computational advantages, we performed a detailed cost analysis comparing FullereneNet with DFT and MLIP approaches. The DFT calculations (using Gaussian 16<sup>37</sup> on 48 CPU cores), geometry optimization with B3LYP/6-31G\* required 33 hours 56 minutes of wall time, followed by an additional 6 hours 4 minutes for single-point energy calculations with B3LYP/6-311G\*, totaling approximately 40 hours of computation time. In contrast, GAP-20 calculation (using LAMMPS with convergence criteria of  $10^{-4}$  eV for total energy and  $10^{-6}$  eV  $\text{\AA}^{-1}$  for atomic forces) completed geometry optimization in approximately 1 minute on a single CPU core. Remarkably, FullereneNet prediction required less than 5 seconds with one NVIDIA L4 GPU, representing a tremendous speedup compared to DFT calculations. This dramatic computational acceleration arises from FullereneNet's ability to directly predict binding energies based on the arrangement of polygonal rings, eliminating the need to compute energies from optimized geometry using DFT or MLIP.

In summary, our results demonstrate that FullereneNet effectively leverages topological features from unoptimized structures to accurately predict binding energies, showcasing strong extrapolation capabilities for larger fullerenes. In contrast, MatFormer and GAP-20 rely heavily on optimized structures for accurate prediction. Given GAP-20's exceptional ability to generate optimized geometries closely matching DFT results, integrating it with FullereneNet could enhance predictions of additional properties, such as ionization potential and electron affinity. This combined approach will be explored further in Section 3.3.

### 3.2 Feature analysis

Feature engineering remains a fundamental challenge in materials and molecules discovery, where the choice of structural representations significantly impacts model performance. For conventional GNN models designed for molecules, researchers typically utilize atom features such as element type and hybridization, along with bond features including bond types and distances.<sup>53</sup> However, these conventional features become inadequate for fullerene modeling, as they consist exclusively of  $sp^2$ -hybridized carbon atoms, resulting in uniform node features across the structure. In this study, we

developed topologically informed node and edge features based on pentagon and hexagon arrangements and investigated whether these features can capture the structural nuances of fullerene systems composed of chemically identical atoms.

As detailed in Section 2.3, all node and edge features were derived from the adjacency matrix, whose elements reflect the connectivity between pairs of carbon atoms within a fullerene molecule (see Fig. 1). Given that the connectivity among carbon atoms varies across different fullerene structures, each yields a distinctive adjacency matrix, thereby enabling a unique representation of each structure. However, when using only the adjacency matrix, along with Gaussian-random-sampled node and edge features as inputs, the GNN model demonstrates extremely low predictive capability (see Fig. S6). These findings indicate that, while the node and edge features are derived from the adjacency matrix, they offer distinct structural dimensions crucial for interpreting structure–property relationships. Specifically, the adjacency matrix records atom-pair connectivity, capturing local connectivity within a fullerene molecule. In contrast, node features specify the types of three rings each atom shares, and edge features detail the types of four rings shared by each bond, providing semi-local structural information that cannot be effectively inferred by the GNN model but must be incorporated through human domain expertise. This strategic integration of chemical knowledge about ring types and arrangements enables our model to differentiate between carbon atoms that would otherwise appear indistinguishable, establishing a hierarchical representation that captures both local connectivity and higher-order topological patterns essential for stability prediction. Our findings highlight the crucial role of human domain knowledge in feature extraction and representation design, contributing to the development of more robust, reliable, and accurate ML models.

The manually derived node and edge features enable the GNN-based model, FullereneNet, to achieve superior performance in binding energy predictions, as demonstrated above. Since both node and edge features are derived from a single adjacency matrix to capture the semi-local chemical environment of pentagons and hexagons, we further evaluated the necessity of incorporating both feature types. To this end, we retrained the FullereneNet model using only node features to



predict C–C binding energies across three test sets. The corresponding performance metrics are presented in Fig. 5 and Table S5. Similar to the model trained with both node and edge features, the model utilizing only node features demonstrates excellent extrapolation performance across four training strategies, yielding an average  $R^2$  of 0.989 and an MAE of 3 meV per atom for  $C_{60}$ , 0.974 and 4 meV per atom for  $C_{70}$ , and 0.364 and 6 meV per atom for  $C_{72}$ – $C_{100}$ . The slightly reduced accuracy for the  $C_{72}$ – $C_{100}$  test set can be attributed to its binding energy distribution, which falls outside the range of the training set (see Fig. S1). These results suggest that since both node and edge features originate from the adjacency matrix, the inclusion of edge features offers minimal additional advantage in enhancing the extrapolation performance of GNN models when node features are already incorporated.

### 3.3 Further discussion

One of the key challenges in the computational design of molecules and materials is the high computational cost associated with structural optimization.<sup>54,55</sup> In previous sections, we demonstrated that the FullereneNet achieves superior performance in predicting fullerene stability through carefully designed topological features rather than relying on optimized 3D Cartesian coordinates. This approach offers a substantial advantage by eliminating the computational cost associated with geometry optimization, which can be prohibitively expensive for high-throughput screening. To reinforce this advantage, we evaluated FullereneNet's predictive capability across a broader range of properties beyond stability. We tested FullereneNet on 11 other essential properties relevant for practical applications,<sup>56</sup> including various electronic characteristics and solubility metrics, as summarized in Table S6.

First, we benchmarked the MatFormer model using both unoptimized and optimized structural data as input. It is important to note that GAP-20 is limited to predicting the stability of fullerenes and cannot forecast other fundamental properties. As shown in Fig. S7, the MatFormer model struggles to accurately capture the structure–property relationships when using unoptimized structures on 11 properties. In contrast, structure optimization leads to significant improvements in both  $R^2$  and MAE values. For example, in predicting the HOMO–LUMO gap, the  $R^2$  value increased from  $-0.64$  to  $0.51$ , while the MAE decreased from  $0.23$  eV to  $0.12$  eV (Fig. S7 and S8).

Subsequently, we retrained the FullereneNet model using both node and edge features derived from unoptimized structures and applied the model to extrapolate predictions for the  $C_{60}$  dataset. As illustrated in Fig. 6 and S8, FullereneNet achieves comparable predictions to MatFormer with optimized structures for electronic properties while outperforming MatFormer in solubility-related property predictions with higher average  $R^2$  and lower MAE values. Specifically, when predicting free solvation energies in water ( $\Delta G_{\text{sol}}(\text{water})$ ) and 1,2-dichlorobenzene ( $\Delta G_{\text{sol}}(\text{ODCB})$ ), and the 1,2-dichlorobenzene–water partition coefficient ( $\log P$ ), FullereneNet achieves MAE values of  $0.80$   $\text{kJ mol}^{-1}$ ,  $0.69$   $\text{kJ mol}^{-1}$ , and  $0.06$ , respectively. In contrast, the MatFormer model trained on optimized structures yielded MAE values of  $0.85$   $\text{kJ mol}^{-1}$ ,  $1.74$   $\text{kJ mol}^{-1}$ , and  $0.26$  (Fig. S8). These results highlight the effectiveness of our feature design in capturing the chemical characteristics of fullerene systems, thereby enhancing the transferability of the FullereneNet model in predicting a diverse range of fundamental properties of fullerenes.

It is important to note that while FullereneNet achieves exceptional accuracy in predicting binding energies ( $R^2 = 0.99$ ),

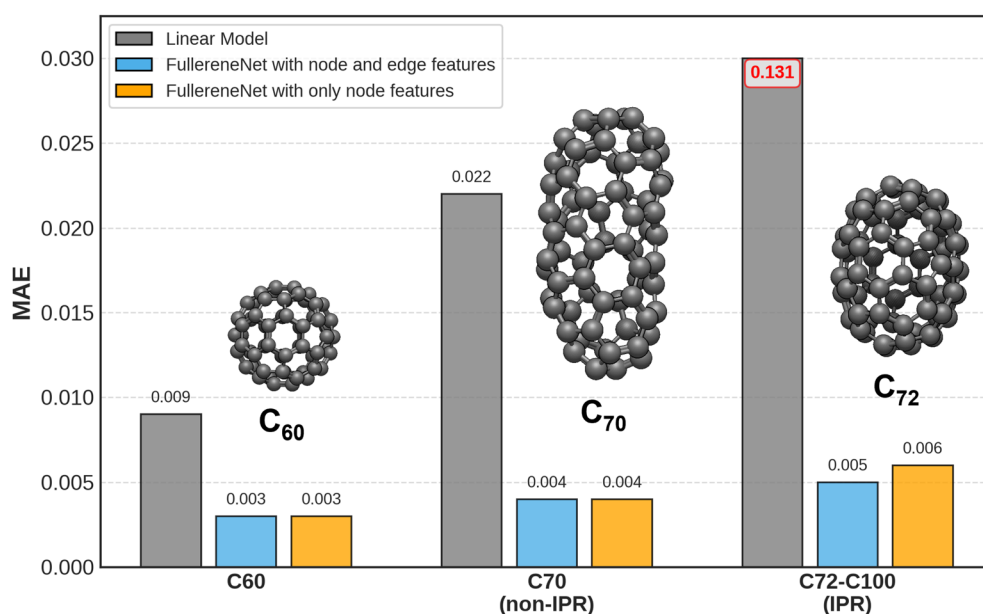


Fig. 5 Comparison of binding energy prediction performance across diverse fullerene datasets. Three models are evaluated: a linear regression model (with notably high error values, shown in red), FullereneNet utilizing both node and edge features, and FullereneNet employing only node features. Inset molecular structures depict three representative fullerenes from our dataset:  $C_{60}$  isomer 1 (left),  $C_{70}$  isomer 11 (center), and  $C_{72}$  isomer 1 (right).



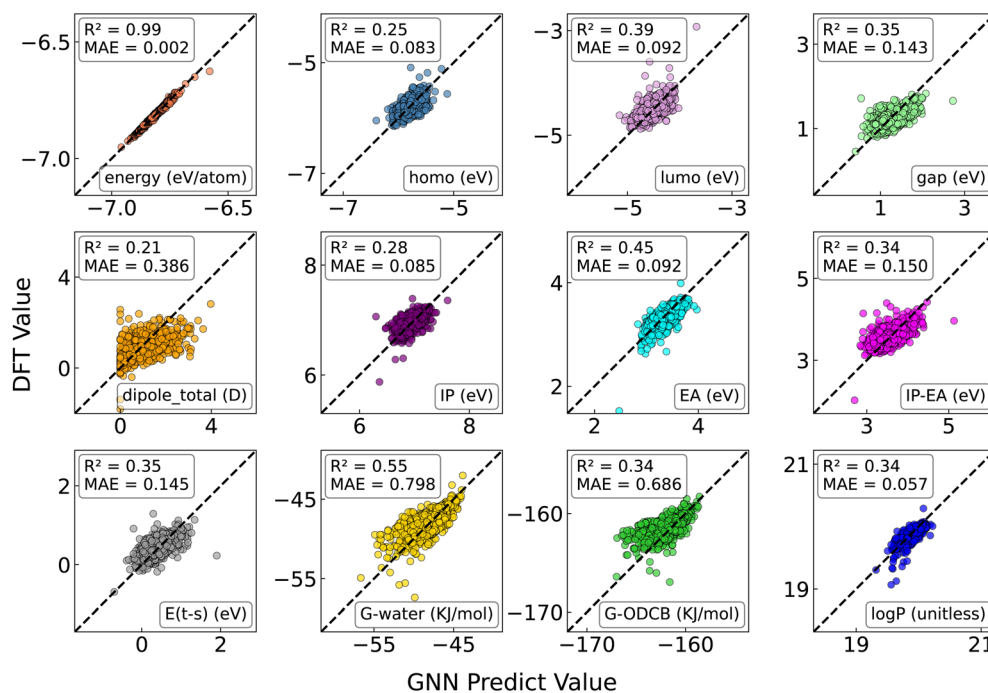


Fig. 6 Extrapolation performance of FullereneNet on 12 properties in the  $C_{60}$  dataset, trained on data from  $C_{20}$  to  $C_{58}$ . The X-axis represents the predicted values of various properties obtained using FullereneNet, while the Y-axis shows the corresponding DFT-calculated values. The labels' meanings for each subplot are summarized in Table S6.

its performance varies across other properties. For instance, electronic properties such as the HOMO–LUMO gap and electron affinity exhibit moderate predictive accuracy ( $R^2 = 0.35$  and  $0.45$ , respectively), indicating that these quantum mechanical properties are influenced by factors beyond the current topological descriptors. This observation is consistent with established chemical principles, as electronic properties often require more sophisticated quantum mechanical descriptors to accurately capture electron density distributions and orbital interactions.<sup>57</sup> Nonetheless, FullereneNet provides reliable predictions across multiple properties without costly geometric optimization, marking a significant advancement in high-throughput fullerene screening. By effectively balancing computational efficiency and predictive accuracy, our model enables the rapid identification of promising candidates for further computational or experimental validation. These findings also highlight the limitations of topological descriptors in capturing complex electronic properties while underscore the broader applicability of FullereneNet in efficient property prediction.

## 4 Conclusion

In this work, we developed a graph neural network (GNN)-based model, FullereneNet, to predict a wide range of fundamental properties of fullerenes using topological features derived from unoptimized structures. By leveraging the chemical environments of pentagons and hexagons within the fullerene cage, we demonstrated that these topological features efficiently capture the local structural details of fullerenes, enabling accurate

property predictions without the need for computationally expensive quantum chemistry optimizations. Our model significantly outperforms existing machine learning interatomic potentials GAP-20 and MatFormer, achieving superior accuracy in predicting C–C binding energies across various fullerene sizes. Additionally, FullereneNet exhibits robust performance in predicting 11 other properties, including HOMO–LUMO gaps, solvation free energies, and partition coefficients, demonstrating its versatility and transferability. This study provides a computationally efficient framework for high-throughput screening of fullerene candidates, offering a valuable tool for advancing the exploration and application of fullerenes in various fields, from optoelectronics to materials science.

## Author contributions

B. L. and M. L. initiated this study. B. L. conducted the theoretical calculations, while J. J. trained the machine learning models. All authors contributed to data analysis and manuscript writing.

## Conflicts of interest

There are no conflicts to declare.

## Data availability

The fullerene structures, including both unoptimized and optimized geometries, are available in the Zenodo repository.<sup>58</sup>



Code availability: the Python implementation of Fullerene-Net is openly accessible on GitHub and Zenodo for long-term availability and reproducibility.<sup>59</sup>

Supplementary information: including details on extension of feature construction method, MatFormer model's details; figures on dataset distribution, model performance on different test sets; tables of fullerene isomer numbers, model parameters, and property units is available. See DOI: <https://doi.org/10.1039/d5dd00241a>.

## Acknowledgements

This work was financially supported by the University of Florida's new faculty start-up funding. The authors acknowledge the University of Florida Research Computing for providing computational resources and support that have contributed to the research results reported in this publication.

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