

Showcasing research from Professor Garcia's laboratory, Institute of Chemical Technology, Universitat Politècnica de València (Spain), and co-authors from Lorestan University (Iran), Universidad Autónoma de Madrid/IAAdChem (Spain), Madurai Kamaraj University (India), and Eskisehir Technical University (Turkey).

Sulfonated porous organic polymers: strategic design, synthesis, and applications in catalysis, adsorption, and energy-related processes

Porous organic polymers having sulfonate groups have emerged as novel functional materials spanning a range of applications from fluorine-free anionic exchange resins and membranes for electrolyzers and fuel cells to solid acids promoting sustainable chemical transformations.

Image reproduced by permission of Hermenegildo García Gómez from *Chem. Soc. Rev.*, 2026, **55**, 2635.

Artwork: Katya Cuevas Bercovich.

As featured in:



See Ali Reza Oveisi, Saba Daliran, José Alemán, Hermenegildo García *et al.*, *Chem. Soc. Rev.*, 2026, **55**, 2635.



Cite this: *Chem. Soc. Rev.*, 2026, 55, 2635

Sulfonated porous organic polymers: strategic design, synthesis, and applications in catalysis, adsorption, and energy-related processes

Ali Reza Oveisi, ^{*a} Saba Daliran, ^{*a} Matías Blanco, ^{bc}
 Amarajothi Dhakshinamoorthy, ^d Unal Sen, ^e José Alemán ^{*bc} and
 Hermenegildo García ^{*f}

Porous organic polymers (POPs) have become an innovative class of tailor-made materials, encompassing a variety of frameworks that range from highly crystalline to fully amorphous structures such as covalent organic frameworks (COFs), covalent triazine frameworks (CTFs), porous aromatic frameworks (PAFs), conjugated microporous polymers (CMPs), polymers of intrinsic microporosity (PIMs), and hyper-cross-linked polymers (HCPs). While their inherent porosity and stability are impressive, the true strength of POPs lies in strategic functionalization. Among the various methods reported, the incorporation of sulfonic acid ($-\text{SO}_3\text{H}$) groups in these porous scaffolds introduces additional functionality. This review explores a comprehensive overview of sulfonated POPs (SPOPs), where robust frameworks are combined with the strong Brønsted acidity of $-\text{SO}_3\text{H}$ groups. We describe the design and synthesis of SPOPs, highlighting how this functionalization tailors their properties for innovative applications. Moving beyond their well-known role as superior heterogeneous acid catalysts for organic transformations, SPOPs are now emerging as key materials for addressing global challenges. Their remarkable capabilities are evident in environmental applications, including their deployment as high-performance adsorbents for the removal of dyes, antibiotics, and heavy metals from water, as well as functional porous solids for selective gas separation. We also explore their pioneering applications as next-generation proton-conducting membranes for high-performance fuel cells and advanced energy storage systems, offering alternatives to fluorinated membranes. This review delivers both a critical analysis of the current state-of-the-art and a forward-looking perspective on the challenges and opportunities ahead, serving as a roadmap for leveraging the multifunctional properties of SPOPs to advance sustainable chemistry, environmental remediation, and energy technologies.

Received 30th October 2025

DOI: 10.1039/d5cs00918a

rsc.li/chem-soc-rev

1. Introduction

Porous organic polymers (POPs) are a relatively new class of soft materials characterized by scaffolds built predominantly from light elements.^{1–3} Typically, C, N, O, B, and H integrate the

structure of the polytopic monomer subjected to the polymerization for yielding this family of materials, but other elements (*e.g.*, S, P) may also be incorporated.^{4,5} The evolution of synthetic porous solids began with zeolites and advanced through metal–organic frameworks (MOFs), which combine metal clusters with organic linkers; the foundational work by Kitagawa,^{6,7} Robson,⁸ and Yaghi,⁹ recognized by the Nobel Prize 2025 in Chemistry, started as early as 1989 and was firmly established in the 1990s. The field later expanded into purely organic frameworks, which are the structures of this review focused on. Indeed, POPs are fundamentally categorized by their structural order into amorphous types, which lack long-range molecular arrangement, and crystalline types, which possess a highly ordered repeating structure. The development of POPs has been marked by several key milestones, all based on how the monomers are assembled by covalent bonds in space.^{10,11} The specific choice of monomer, linkage chemistry,

^a Department of Organic Chemistry, Faculty of Chemistry, Lorestan University, Khorramabad 68151-44316, Iran. E-mail: oveisi.a@lu.ac.ir, daliran.s@lu.ac.ir

^b Organic Chemistry Department, Módulo 1, Universidad Autónoma de Madrid, 28049 Madrid, Spain. E-mail: jose.aleman@uam.es

^c Institute for Advanced Research in Chemical Sciences (IAChem), Universidad Autónoma de Madrid, 28049 Madrid, Spain

^d School of Chemistry, Madurai Kamaraj University, Madurai, 625021, Tamil Nadu, India

^e Department of Materials Science and Engineering, Faculty of Engineering, Eskisehir Technical University, Eskisehir 26555, Turkey

^f Instituto Universitario de Tecnología Química, (CSIC-UPV), Universitat Politècnica de València, Valencia 46022, Spain. E-mail: hgarcia@qim.upv.es



and polymerization conditions dictates the final material's structure, leading to several distinct classes (Fig. 1). Under this general premise, the foundational work on hyper-cross-linked polymers (HCPs) dates back to the 1970s,¹² pioneered by Davankov and colleagues,^{13–15} who created networks with permanent microporosity. The field expanded significantly in the 2000s with the introduction of new material classes. Polymers of intrinsic microporosity (PIMs) were first reported in 2002, emerging from the successful preparation of phthalocyanine-based network polymers (Pc-Network-PIMs).¹⁶ A key advancement followed in 2004 with the development of processable PIMs, distinguished by their rigid, contorted backbones that generate intrinsic micropores.^{17–19} This was followed by the development of first crystalline covalent organic frameworks

(COFs) in 2005,^{3,20,21} which demonstrated the potential for designing ordered, covalently bonded structures with predictable topology.^{22–25} The subsequent years, beginning in 2007,²⁶ witnessed the concurrent development of conjugated microporous polymers (CMPs)^{26–29} and further advancements in HCP synthesis.³⁰ Particularly significant subclass, covalent triazine frameworks (CTFs),³¹ was introduced around 2008 and is defined by nitrogen-rich, robust aromatic triazine linkages.^{32–36} While traditional CTFs synthesized *via* irreversible routes are often amorphous, further advanced strategies have yielded crystalline versions.^{37–39} Historically, due to their π -conjugated skeletons, some amorphous CTFs were grouped under the broader category of CMPs, a class known for combining permanent nanoporosity with optoelectronic properties. The family of porous



Ali Reza Oveisi

Ali Reza Oveisi is an Associate Professor of Organic Chemistry at Lorestan University. After earning his PhD from Bu-Ali Sina University (2015) and conducting research as a visiting scholar in the Hupp and Farha groups at Northwestern University, he founded his independent research group at the University of Zabol in 2017. His work focuses on designing and functionalizing advanced porous materials, including MOFs, COFs, and POPs, for applications in catalysis, photocatalysis, sustainable fuel production, and environmental remediation. His research has been recognized with multiple Scientist Awards, and he currently serves as a Guest Editor for a Jove Special Issue.



Saba Daliran

Saba Daliran holds a BSc from Lorestan University (LU) and an MSc/PhD from Bu-Ali Sina University. She was a visiting scholar with Prof. Hermengildo Garcia at Universitat Politècnica de València and completed NSF- and ISEF-funded postdocs. Now an Assistant Professor at LU, she focuses on (nano)porous materials. Her honors include the Dr Kazemi Ashtiani Award, recognition as a Distinguished Researcher of Lorestan University and Province, and selection among Iran's "Nation's Scientific Elites" (2025).



**Amarajothi
Dhakshinamoorthy**

Amarajothi Dhakshinamoorthy received his PhD degree in 2009 from Madurai Kamaraj University, Madurai-21, India. Later, he had his postdoctoral stay with Prof. H. Garcia at the Technical University of Valencia for four years. Currently, he is serving as UGC Assistant Professor at the School of Chemistry, Madurai Kamaraj University, India. His research interests include catalytic/photocatalytic applications of MOFs and graphene-related materials and MOFs for energy applications. He is the recipient of the Young Scientist Award 2014 for Chemical Sciences by The Academy of Sciences, India. He has co-authored over 220 international publications, five book chapters, and a patent. He is a Fellow of Royal Society of Chemistry.



Unal Sen

Unal Sen earned his BSc in Chemistry from Bilkent University and his PhD in Engineering from Tohoku University. Following postdoctoral research at Northwestern University under Professors Hupp and Farha, he joined Eskisehir Technical University's Department of Materials Science and Engineering. His research centers on MOFs, MOF-derived carbons, and MOF-polymer hybrids for applications in energy storage, gas separation, environmental remediation, and advanced ceramics. His contributions have been recognized with a Japan Society for the Promotion of Science RONPAKU Fellowship.



aromatic frameworks (PAFs) was notably advanced in 2009 with the report of PAF-1,⁴⁰ a prototype material exhibiting an exceptionally high surface area.^{41,42} The term “aromatic” in PAFs originates from the abundant benzene rings that form the core building blocks of this and other frameworks in the family.^{41–44} This progression has yielded a diverse family of materials prized for their high surface areas, tunable porosity, and exceptional stability.

The main differences among these porous polymers are crystallinity, the type of chemical bonds, and the specific building blocks used, with the choice of reaction directly influencing the resulting polymer (Table 1).

A decade of rapid innovation has significantly advanced the field of porous organic materials. This has led to the reporting of diverse families, known under various general nomenclatures such as POPs,^{2,47} porous organic frameworks (POFs),^{48,49} covalent organic polymers (COPs),^{50–52} porous polymer networks

(PPNs),^{53–55} microporous organic networks (MONs),⁵⁶ and porous organic networks (PONs).^{57,58} While nomenclature can vary, POPs are often used as an overarching term for this class of materials, encompassing both crystalline and amorphous frameworks with entirely organic, covalently bonded structures. These materials are typically characterized by high surface areas and permanent porosity, and their applications in gas storage,^{54,59–61} adsorption/separation,^{62–71} and catalysis have been extensively investigated.^{72–83}

POPs share several defining characteristics. Their high surface areas stem from deliberate synthetic design,^{84,85} which creates an interconnected network of monomers and dictates pore size and distribution.⁸⁶ Synthetic design and conditions may also induce and/or influence the crystallinity degree of the material.^{87,88} In addition, they present usually semiconducting behavior since most of them contain aromatic patches that allow charge separation and carrier mobility.^{1,89} Despite being composed of light elements, they often possess good mechanical properties⁹⁰ and demonstrate notable chemical and thermal stability.^{91–93} As a whole, the general properties make POPs suitable candidates for advanced applications in (opto)-electronics,^{89,94} gas separation,⁹⁵ catalysis,² environmental remediation⁶³ and transformation of renewable feedstocks,⁹⁶ CO₂ capture and fixation,⁹⁷ bioapplications,⁹⁸ and energy storage and conversion, among others.^{62,93,95,99–114} However, the properties of POPs can be further enhanced and tailored for specific applications through post-synthetic functionalization.¹¹⁵ An important functionalization of POPs, as discussed below, is based on the introduction of sulfonic and/or sulfonate sites. By introducing –SO₃H groups, and maybe further reactivity with them to yield sulfonates, the properties of the POP material are completely changed due to the incorporation of a highly acidic (or, through proton-metal exchange, a metal-chelating group) in the polymeric structure.¹¹⁶ SPOPs offer compelling advantages over conventional acid systems. Compared to homogeneous strong acids like H₂SO₄ or *p*-toluenesulfonic acid (PTSA), which pose challenges in separation, recycling, and corrosion, SPOPs provide comparable acid



Hermenegildo García

Hermenegildo García is a Full Professor at the Instituto de Tecnología Química of the Technical University of Valencia, Spain. His research focuses on heterogeneous catalysis using porous materials and the photocatalytic production of solar fuels, with over 1000 published articles. His distinctions include the Spanish National Research Award (2021), the Jaume I Prize for Novel Technologies (2016), and being named a Doctor

Honoris Causa by the University of Bucharest. He is also the recipient of the 2011 Janssen-Cilag award and was a 2021 Medal Lecturer for the International Association for Advanced Materials.



José Alemán (left) and Matías Blanco (right)

Dr Matías Blanco (right) is Assistant Professor at UAM and contributes his research expertise in the field of catalytic materials. He obtained his degree and PhD from the University of Oviedo and specializes in materials science, combining surface science with the development of heterogeneous catalysts to enhance organic transformations through supramolecular and covalent surface chemistry. His interests in carbon nanomaterials and porous polymers further strengthen the multidisciplinary vision of Alemán's group.

José Alemán (left) defended his doctoral thesis in 2006 and was promoted to Full Professor in 2023. He has received numerous prestigious awards, including the Lilly Prize for Best Doctoral Student, the UAM Best Doctoral Thesis Award, the Sigma-Aldrich Award for Young Researchers (RSEQ), the Lilly Young Researcher Award, the José Barluenga-RSEQ Medal, and the Ignacio Rivas-RSEQ Medal. His research focuses on asymmetric catalysis and catalytic materials. He has authored 240 scientific publications, supervised 28 doctoral theses and more than 40 Bachelor's and Master's theses, and led 18 competitively funded research projects. He currently serves as Director of the Advanced Institute of Chemical Sciences at UAM, Vice President of the Organic Chemistry Division of the Royal Spanish Chemical Society, and is a member of the Royal Academy of Exact, Physical and Natural



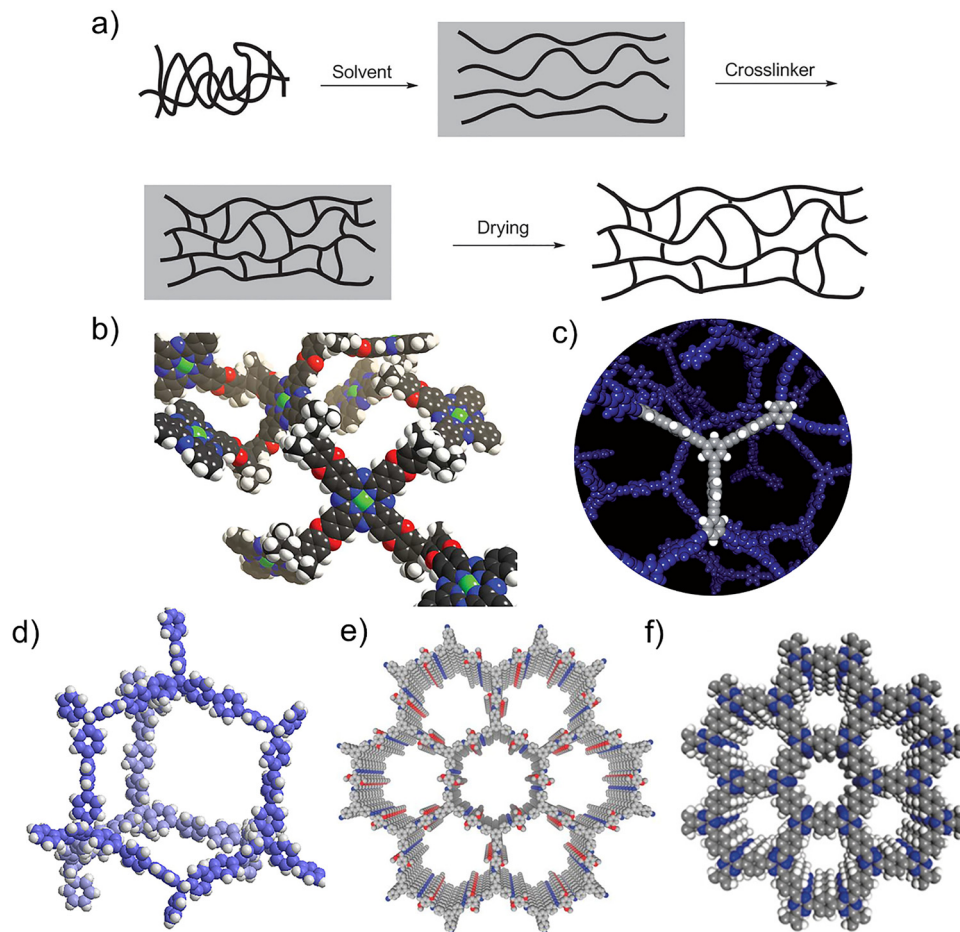


Fig. 1 Different types of POPs: a) HCP, Reproduced with permission from ref. 45 Copyright 2007 Royal Society of Chemistry; b) PIM, Reproduced with permission from ref. 17 Copyright 2004 Royal Society of Chemistry; c) CMP, Reproduced with permission from ref. 26 Copyright 2007 Wiley-VCH; d) PAF, Reproduced with permission from ref. 43 under a Creative Commons CC BY 4.0 license, 2016 MDPI; e) COF, Reproduced with permission from ref. 46 Copyright 2015 Nature Publishing Group; f) CTF, Reproduced with permission from ref. 35 Copyright 2018 Wiley-VCH.

Table 1 Summary comparison of the main features of POP materials

POP material	Crystallinity	Key characteristics & bonding	Typical reactions/formation
COFs	Crystalline	Designable, ordered structures with high surface area, defined pore size, and reversible synthetic reactions	Reversible reactions include those forming boron-based linkages (<i>e.g.</i> , boronate ester, boroxine, borazine), imine-based linkages (<i>e.g.</i> , imine, hydrazone, azine, β -ketoenamine), and other dynamic linkages such as squaraine bonds and keto-enol tautomerism
HCPs	Amorphous	Inexpensive and scalable	The most common method for irreversible cross-linking is Friedel–Crafts (FC) alkylation
CMPs	Amorphous	Extended π -conjugation systems are associated with optoelectronic properties	Irreversible metal-catalyzed coupling reactions, such as Suzuki, Sonogashira, Scholl, and Yamamoto coupling, as well as Buchwald–Hartwig amination, and (less commonly) Schiff base formation
PAFs	Amorphous	Exceptional surface area and stability, characterized by rigid aromatic frameworks	Irreversible cross-coupling reactions, including Yamamoto-type Ullmann, Suzuki, and Sonogashira–Hagihara couplings, and nucleophilic substitution reactions
CTFs	Mostly amorphous, though crystalline forms are possible	Rich nitrogen content (triazine units), high chemical stability	Nitrile or ketone trimerization utilizing ionothermal synthesis, P_2O_5 catalysis, and superacid catalysis

strength while enabling easy recovery and reuse as heterogeneous solids. This merges high activity with improved process sustainability. Furthermore, SPOPs surpass classical polymeric

acid membranes¹¹⁷ (*e.g.*, Nafion[®], sulfonated poly(aryl ether)s) by leveraging intrinsic, engineered nanoporosity. Unlike the often random pore structures of traditional materials, the



well-defined, tailorable architecture of SPOPs offer superior control over acid-site density, pore environment, and surface chemistry. This precise engineering enhances mass transport, substrate accessibility, and performance in catalysis, separation, and proton conduction. Consequently, SPOPs are uniquely suited for green chemistry challenges. Their combination of strong solid-state acidity, robust frameworks, and tunable porosity makes them ideal catalysts for transforming renewable feedstocks and for critical carbon management processes, including CO₂ capture, fixation, and conversion. Additionally, the -SO₃H groups facilitate proton-electron conductivity, opening avenues in advanced energy technologies such as fuel cells and batteries. By integrating these functionalities within a designed porous architecture, SPOPs overcome the limitations of traditional solid acids and emerge as flagship materials for sustainable chemical processes and next-generation applications. The development and use of SPOPs have followed a unique evolutionary path since their early beginnings, summarized in Fig. 2 and discussed in the following paragraphs.

It is important to note that efficient solid acid catalysts synthesized *via* routes such as the sulfonation of swelling mesoporous polydivinylbenzenes (DVBs),¹¹⁸ pristine polymers often prepared through radical polymerization, were reported prior to this period, highlighting an active parallel research avenue. However, the modern scope focused on diverse synthetic strategies and tailored functionalities gained significant momentum with a key report in 2011, where a PPN was used for selective CO₂ adsorption, marking the first exploration of functional POPs in separation science (Fig. 2).⁵³ This key work opened the door for more diversity, and by 2015, HCPs¹¹⁹ and COFs¹²⁰ were reported as effective solid acid catalysts for biofuel production, thus broadening their applications to catalysis.¹²¹ At the same time, early sulfonation techniques started to be systematically studied to improve polymer functionality. The period from 2016 to 2020 witnessed a significant growth in both material families and their applications. In 2016, PAFs¹²² and related POPs were used as highly effective catalysts.¹²³ In the same year, POPs, including COFs,¹²⁴ also demonstrated promising proton conductivity, showcasing their increasing versatility in energy-related uses.¹²⁵ This trend continued with CMPs¹²⁶ and CTFs¹²⁷ being studied as

efficient proton conductors by 2018–2020. Additionally, the range of applications expanded beyond energy and catalysis, as seen in 2019 when sulfonated MONs were successfully used for drug delivery, highlighting their potential in biomedical and environmental fields.⁵⁶ Since 2020, the field has entered a maturation phase focused on performance optimization and the development of advanced, diverse applications in catalysis, environmental remediation, and energy technologies.^{99,116,125,127–133} Major breakthroughs include the engineering of PAFs for rapid Li-ion conduction,¹³⁴ significant improvements in gas separation properties in sulfonated PIMs,^{135,136} and ongoing efforts to develop advanced membranes for fuel cells and high-performance proton conductivity.^{137–139} This ongoing timeline highlights continuous and increasingly rapid research efforts, showing that innovation within the SPOPs family, spanning a diverse range of architectures, for vital roles in separation, catalysis, energy, and environmental science, remains very active and continues to evolve without reaching a limit.

This review focuses on those modified materials, SPOPs: how they are synthesized and/or post-synthetically modified, which new properties arise, and how they govern the application for which the material has been selected. We can anticipate that the defining feature of SPOPs is their strong solid-state acidity. Indeed, the -SO₃H is one of the strongest organic acids,¹⁴⁰ and in particular, a wide variety of systems typically employ PTSA when a strong organic acid is needed to increase proton concentration and then, trigger the intended phenomenon.^{141–145} Classically, an acid is a substance that dissociates in solution to donate a proton (H⁺), while a base accepts this proton (Scheme 1a). The strength of an acid is quantified by its pK_a value (the negative logarithm of its dissociation constant). The sulfonic group is a strong acid due to the high electronegativity of the S(vi) center, which is bonded to multiple oxygen atoms, combined with the resonance stabilization of the resulting sulfonate conjugate base (Scheme 1b). Thus, PTSA exhibits a pK_a value of -2.8,¹⁴⁶ indicating a very strong acid character considering the organic nature of the molecule. When the -SO₃H is heterogenized on a platform like POP, each repeating unit in the polymer contains a sulfonic moiety, and therefore, the density of acidic groups per mass of polymer is

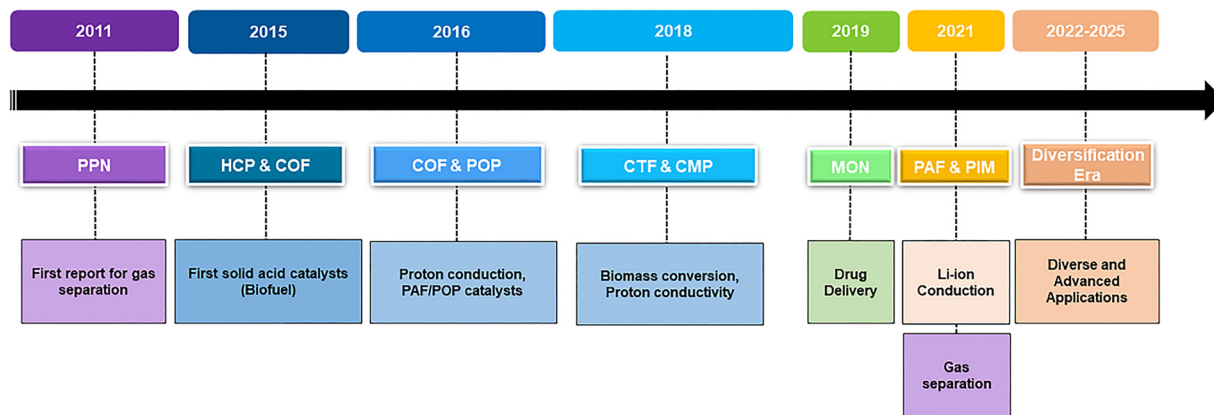
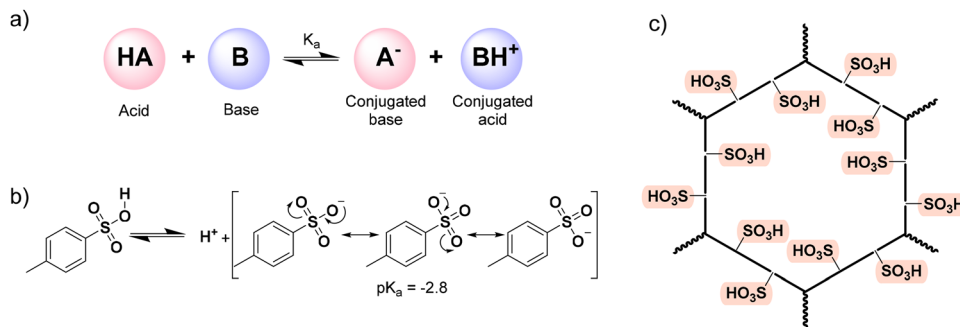


Fig. 2 Evolutionary timeline of SPOPs, showing the development of key material families and their expanding applications from 2011 to 2025 toward future advanced applications.





Scheme 1 Acidity with POPs. a) general acid–base reaction; b) dissociation equilibrium of PTSA; c) general representation of a sulfonated polymer.

large (Scheme 1c).¹⁴⁷ Therefore, a $-\text{SO}_3\text{H}$ functionalized POP might be considered as a solid acid: an acid insoluble in the reaction medium.¹⁴⁸ Since it is insoluble, pK_a cannot be determined, and the acid character is only measured with the facility of dissociating the O–H bond and releasing H^+ units. Nevertheless, it has to be considered that, when heterogenized, the sulfonic functions are attached to solid support, thus diffusion and surface effects, as well as parallel interactions with other functional groups found in the scaffold of the POP, may occur, which can diminish or enhance the acidic character. Furthermore, the heterogenized sulfonic group can also serve for further post-modification or even metal coordination to generate different but relevant applications apart from acidity. Summing up, the feature of a strong heterogenized acid on the structure of a POP supposes an advantage compared to classic systems. Indeed, SPOPs merge tunable reactivity with the inherent ease of separation of a heterogeneous material, allowing for simple recovery (*e.g.*, *via* filtration or centrifugation) and thereby enhancing process efficiency and sustainability.

As a whole, this review will discuss the various methods for the design of sulfonic/SPOPs, how the presence of the $-\text{SO}_3\text{H}$ units affects the structure and the properties of the material, including robustness, shape, size, and crystallinity, and then, the most relevant applications in various fields such as catalysis, environmental remediation, gas separation, and energy-related applications (*e.g.*, proton conductivity, fuel cells, and batteries).

2. Sulfonic/sulfonated POPs

2.1. Design and modification strategies of SPOPs

The POPs literature shows that the most prevalent methods for the material synthesis are a collection of reactions, most commonly based on reversible processes, though irreversible and photochemical routes are also employed.^{149–152} While new reactions are continually emerging, the most widely employed polymerization methods include boronate/boroxime chemistry, the Schiff base condensation (imine, hydrazone, β -ketoenamine, azine, or oxime among others), carbonyl condensation (amidation, Knoevenagel reaction, imide, phenazine and (thio)imidazole ring fusion among others), olefin metathesis, Scholl reaction, FC alkylation, cross coupling methodologies

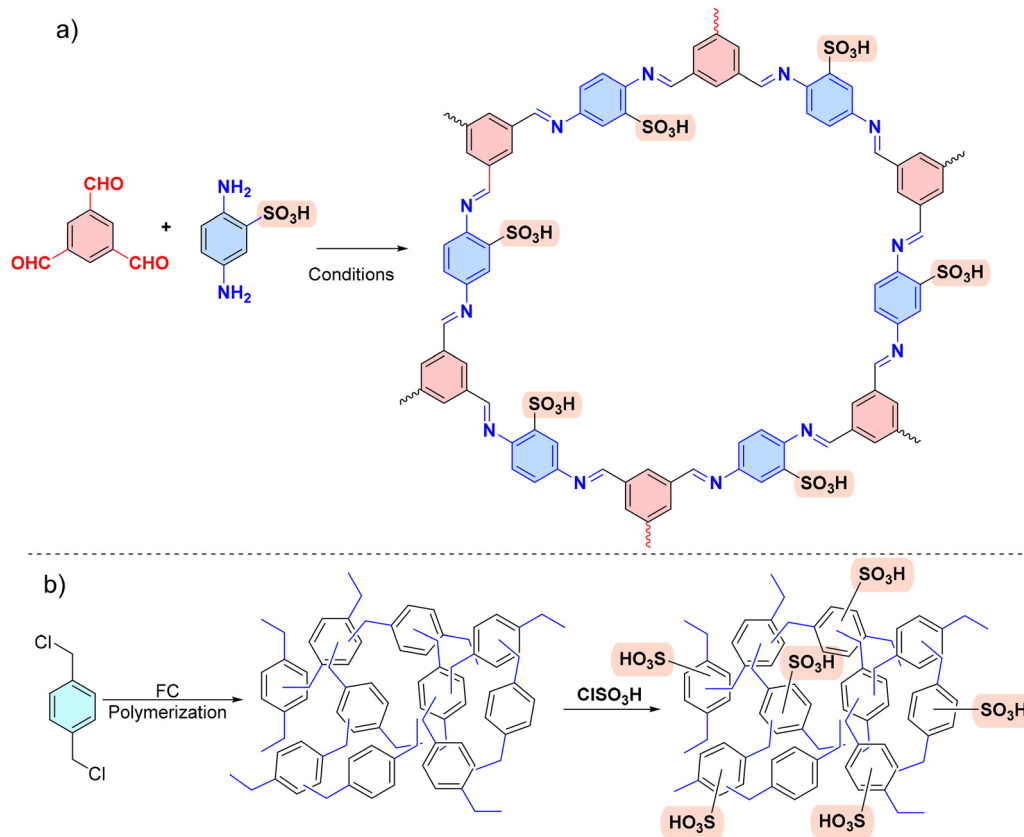
and nitriles trimerization.^{2,23,42,153–155} This diversity results in a wide gamut of structures with tunable properties.

To introduce a $-\text{SO}_3\text{H}$ group, SPOPs have been prepared by two different approaches (Scheme 2a). On the one hand, the $-\text{SO}_3\text{H}$ group can be present in one of the constituting monomers, and the polymerization is performed in its presence without affecting its outcome. As a result, the sulfonic groups of POPs are topologically present by design in each repeating unit. A typical example encompasses the monomer 2,5-diaminobenzenesulfonic acid,¹²⁰ which can be submitted to Schiff base condensation polymerizations with different electrophiles, but other possibilities are reported too.¹⁵⁶ On the other hand, the $-\text{SO}_3\text{H}$ is introduced in a post-synthetic modification (PSM) strategy. Once again, two different reaction pathways are reported. First, the sulfonic group could be introduced using a molecule that directly reacts with the POP core structure, and therefore it is finally fixed to the material network. The most typical strategy employs the aromatic cores of the building blocks to construct the polymer skeleton *via* FC alkylation, followed by sulfonation with ClSO_3H (or SO_3 in some cases), especially in HCP modification. However, the number of sulfonic acids in the polymer structure compared to the previous one is not controlled. Indeed, a typical representation is shown in Scheme 2b, where the bond between the aromatics and the sulfonic group is not (or cannot be) precisely determined. The other type of strategies employs pending functional groups present in the “pristine” polymer that can be transformed into the $-\text{SO}_3\text{H}$. In this case, pending alcohols or amines, including those linking aromatic knots, and even imidazolic units, to be alkylated with sultones, can be employed.^{132,133,157} As a whole, PSM protocols allow the sulfonation of polymers that cannot host the sulfonic group in the monomers, thus expanding the number of systems able to present a sulfonic/sulfonated moiety.

2.2. Effects of the sulfonic group on the POP properties

Studies on SPOPs typically include a control sample without sulfonation for comparison. Both types of samples have been characterized in the same way and have been tested in the same application, usually highlighting the advantages and disadvantages of the functionalization with $-\text{SO}_3\text{H}$. Therefore, a straightforward comparison can be made. In general, the routine characterization of both “pristine” and sulfonated sample





Scheme 2 Different approaches for the synthesis of SPOPs. a) Monomer containing the sulfonic group; b) PSM strategy on HCPs based on FC alkylation.

informs about the presence of the sulfonic group by a series of techniques like Fourier-Transformed infrared (FT-IR) spectroscopy (stretching vibrations of the sulfonic group bonds at ~ 1200 , ~ 1000 and ~ 625 cm^{-1}),¹⁵⁸ thermogravimetric analysis (TGA) with slightly higher weight losses as a result of the presence of the sulfonic moiety,¹⁵⁹ X-ray photoelectron spectroscopy (XPS) with the detection and analysis of S 2p XPS core level region (similar results are obtained by energy dispersive X-ray spectroscopy (EDX), which can be done in combination with a microscopic technique),¹⁶⁰ by wettability tests because the sulfone is more polar¹⁶¹ and increases contact angle, and by electrochemical tests related to the presence of surface acids, such as electrochemical impedance spectroscopy. Very frequently, the amount of $-\text{SO}_3\text{H}$ available for the required application is determined by either elemental analysis or titration measurements.¹⁶² Indeed, HCl and NaOH with adequate indicators in the presence of the POP are employed to determine the acidity of the material.¹⁶³

When the presence of the sulfonic functionality is confirmed, other tests have to be carried out to ensure that the POP structure is preserved after sulfonation. Thus, the assembly of the material is analyzed by solid-state nuclear magnetic resonance (SS-NMR) to check on the aromatic core signals.¹³⁸ Generally, the sulfonation does not strongly modify the skeleton of the POP, and a similar set of signals is frequently achieved. In addition, the crystal structure is usually maintained after the sulfonation. Indeed, powder X-ray diffraction (PXRD) patterns tended to be almost identical,¹⁵⁸ although

some loss of crystallinity can be reported in the shape of less intense and broader peaks since the treatment somehow may affect POP structure and ordering.¹⁶⁴ Finally, the textural properties are checked too, and generally, slightly smaller surface areas with less gas adsorption are obtained as a result of the sulfonation,¹⁶⁵ undoubtedly assigned to the presence of bulky chemical groups in the pore that reduce the porosity of the function material, although in some cases the opposite trend has been observed.¹⁶¹

The final check usually done is how the sulfonation has affected the shape, particle size, and morphology of the samples, and this task is often carried out by microscopic analysis. While the aspect of the materials depends on the polymerization conditions, and very different configurations might be obtained as a function of the reaction conditions and monomers structures, some trends are observed in the literature. In general, the shape and morphology were nicely maintained. For example, the rod-like structure on the material presented by Yang *et al.* had the same aspect both at the scanning and transmission electron microscopes (SEM and TEM).¹⁶¹ Amorphous lamellar structures consistent in shape and lateral dimensions were achieved by Wolska and collaborators, even considering that they yielded HCPs with an increased degree of sulfonation using ClSO_3H .¹⁶⁶ However, very strong functionalization treatments may result in the partial collapse of the structure, as resembled by the microscopic analysis reported by Dong and coworkers.¹⁶⁷ Therefore, the sulfonation must be



performed with some caution, because not the highest sulfonic content would lead to the best material, as also shown in the application sections.

3. Catalysis

The strategic incorporation of $-\text{SO}_3\text{H}$ groups into POPs creates a versatile class of solid Brønsted acids. SPOPs transcend the limitations of conventional solid acids (*e.g.*, Amberlyst-15, Nafion[®]) and homogeneous analogues (*e.g.*, H_2SO_4 , PTSA) by combining four key design principles: (i) high and tunable acid-site density anchored in a robust framework, (ii) engineered nanoporosity for enhanced mass transport and substrate preorganization, (iii) synergistic multifunctionality by co-incorporating other active sites, and (iv) tailored surface chemistry (*e.g.*, hydrophobicity) to optimize the reaction microenvironment. This section will cover those relevant examples where SPOPs are employed as heterogeneous catalysts in different chemical reactions, the main characteristics and catalytic parameters are summarized in Table 2.

3.1. General solid acid behavior in catalysis

The synthesis of POPs where the constituting monomers possess $-\text{SO}_3\text{H}$ yields materials in which the topological pores are intrinsically decorated with sulfonic moieties. Therefore, the number of available active sites is remarkably high, and such materials are generally regarded in the literature as exhibiting strong acidic behavior. On the other hand, PSMs over the aromatic networks (such as the sulfonation with ClSO_3H) or using *ad hoc* functional reactive points also cover the material with $-\text{SO}_3\text{H}$ groups.^{135,153,187} These properties have been harvested for different purposes, for example, the construction of membranes for fuel cells,¹³⁹ as proton-conductive materials,¹²⁴ for adsorption of dyes,¹⁵⁹ or in electrochemical devices, such as desalination, batteries, and supercapacitors.^{203,204}

In catalysis with organic molecules, SPOPs have found application as catalysts in reactions where an acid is needed. Their appeal lies in high surface areas, robust stability, and tunable functionality through monomer selection and/or PSMs like sulfonation. This is a common advantage across POPs. Indeed, this versatility, combined with the cost-effectiveness of subclasses like HCPs, makes this class of materials highly suitable for a diverse range of acid-catalyzed reactions. The intimate link between porosity, acid-site density, and spatial confinement frequently leads to catalytic performances that surpass conventional solid acids (*e.g.*, Amberlyst-15, Nafion[®]) and even rival homogeneous analogues (*e.g.*, H_2SO_4 , PTSA) in certain reactions. The foundational work by Li *et al.* on sulfonated HCPs demonstrated that a balance between acid density and preserved microporosity were key factors to achieve complete conversion and selectivity in FC alkylation and the Beckmann rearrangement (Fig. 3).¹⁶⁸ The material achieved a turnover number (TON) of 43 and demonstrated excellent recyclability, benchmarking favorably against both homogeneous H_2SO_4 and solid acid references (Amberlyst-15, SBA-15-S) (see Table 2, entry 1). This performance

directly highlights the dual advantage of the SPOP as a robust, porous heterogeneous acid. The pore structure connected to the acid moieties was concluded to be responsible for the observed performance as plausible explanation of the observed behavior. In a related study, Sun *et al.* developed a strong acidic SPOP ($H_0 \leq -12.4$) by sulfonating a fluorinated POP, where the electron-withdrawing backbone enhanced $-\text{SO}_3\text{H}$ acidity.¹⁶⁹ This material showed exceptional activity in FC acylation of benzyl alcohol (see Table 2, entry 2). Under continuous flow conditions in a fixed-bed reactor, it maintained stable performance for over 150 min, achieving a steady product yield of approximately 97%. In sharp contrast, commercial solid acids such as Amberlyst-15 and Nafion[®] NR50 deactivated rapidly, falling below 5% conversion after 180 min. By 300 min, FPOP- SO_3H had produced 1.83-, 91.3-, and 131.7-fold more product than POP- SO_3H , Amberlyst-15, and Nafion[®] NR50, respectively. This marked stability is attributed to a synergistic combination of hierarchical porosity and fluorine incorporation. The interconnected pore structure facilitates efficient mass transport, limiting pore blockage and diffusional limitations. Furthermore, as observed in systems like Nafion[®], the electron-withdrawing fluorine atoms enhance both the thermal stability and the acid strength of the $-\text{SO}_3\text{H}$ groups. Together, these features, strong acidity, stabilized active sites, and optimized transport pathways, establish FPOP- SO_3H as a leading porous solid-acid catalyst for this transformation, rivaling state-of-the-art benchmarks. This is a premier example of how the inherently rigid, non-swelling architecture of SPOPs overcome a major limitation of polymeric resin catalysts, enabling practical application in continuous processes. Later, Zhao and coworkers were able to open styrene oxide under acid conditions using a double-decker silsesquioxane-based polymer synthesized *via* a Scholl coupling reaction, which was sulfonated after synthesis in similar strategies than previous polymers.¹⁷⁰ The reaction was completed with perfect activity (full conversion and selectivity in 30 min at room temperature, including a flow chemistry scaling set-up), and the good results were attributed to the high loading of sulfonic acid moieties although the activity was reduced to 50% conversion after the 4th cycle, likely due to the leaching of $-\text{SO}_3\text{H}$ groups into the alcohol medium. Notably, its performance surpassed that of the benchmark solid acid catalyst MIL-DABCO- SO_3H .²⁰⁵ The acid character of this family of materials was employed as a catalyst to construct an acetal from glycols and acetylacetones using a sulfone-decorated polydivinylbenzene-based POP with yields above 95% in just 2 h of reaction.¹⁷¹ This reaction is typically carried out with PTSA in homogeneous phase, therefore the advantage of an easy separation catalyst that increases the overall reaction efficiency is highly beneficial. Yevad *et al.* prepared highly porous HCPs that were sulfonated in a second step using as monomers a mixture of DCX, 2,2',6,6'-tetramethylbiphenyl (MeM) and 2,2',6,6'-tetramethoxybiphenyl (OMeM) (Table 2).¹⁷² These materials were able to catalyze a different condensation reaction that, in this case, yielded diphenylbenzopyrans in numbers higher than 90% at room temperature after 2 h. Moreover, they also prepared triazoles from nitroalkenes and azide excellent yields (exceeding 90%) at room temperature within 1 h. In summary, material synthesis





Table 2 General characteristics and catalytic parameters of SPOPs applied as catalysts

POP	BBs/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
HCP	1,1'-dichloroethylene (DCX) or 4,4'-bis(chloromethyl)-1,1'-biphenyl (BCMP)	FC alkylation with ClSO ₃ H	1335–1900/0.85–2.2	FC alkylation of anisole and benzyl alcohol and Beckmann rearrangement of cyclohexanone oxime	100% conv. and 100% sel.	0.65–2.41	TGA	No	168
HCP	2,2',4,4',6,6'-Hexafluoro-3,3'-divinyl-1,1'-biphenyl	FC alkylation with oleum	340/0.3	FC acylation of benzoyl chloride and anisole	95.3% conv. and >99.5% sel.	3.56	TGA, N ₂ adsorption,	Yes, flow chemistry	169
Double decker type	Phenyltrimethoxysilane	FC alkylation with ClSO ₃ H	9–590/1.52	Epoxide opening	100% conv. and 100% sel.	0.783–1.844	TGA	No	170
Slisesquioxane HCP	Divinyl benzene (DVB)	FC alkylation with ClSO ₃ H	816–1271/30–70	Acetal formation	99.5% conv. and 96% sel.	2.5–3.0	No	No	171
HCP	DCX, 2,2',6,6'-tetramethylbiphenyl (MeM) and 2,2',6,6'-tetramethoxybiphenyl (OMeM)	FC alkylation with ClSO ₃ H	766–1277/1.4–4.8	Synthesis of diphenylbenzopyrans/Naphthopyrans and triazoles	Large scope, yields >90%	4.81% (Elemental analysis)	TGA	No	172
HCP-BBA-0.5	Benzene and benzylamine	PSM	604/ <2, 2.2 and 33	Oil shale upgrading (Catalytic cracking)	Shale oil yield: 41.85%,	2.69	—	—	173
PDVB-SO ₃ H-IS	DVB and sodium <i>p</i> -styrene sulfonate (SPSS/SBS)	<i>De novo</i>	650/ average 11	BisphenolA synthesis	Conv.: ~30%	1.58	Reusability test	To	174
SHCPs (SHCP- <i>x</i> = 1–10)	Bis(chloromethyl) biphenyl	One-pot	540–1060 (micro-macropores)/-	Cyclohexyl acetate hydrolysis	Conv.: up to 75%	0.03–3.76	Reusability test, EA, XPS, TGA	120 mmol	154
P(QP-SBA)-3	SPSS/SBS and phosphonium-based monomer (PQ)	<i>De novo</i>	600/-	Cyclohexyl acetate hydrolysis	Conv.: 93%	3.6	Reusability test	—	129
COF	1,3,5-triformyl phloroglucinol (TFPG) and 2,5-diamino benzenesulfonic acid (DABSA)	<i>De novo</i>	230 (micro-macropores)/-	CO ₂ capture solvent regeneration	Yield: 89%	0.185	TGA	No	175
[DBUH] ₂ Cys@-COF-3	1,8-diazabicyclo[5.4.0]undec-7-ene (DBU) and <i>l</i> -cysteine (Cys) ionic liquid (IL) into TpPa-COF-SO ₃ H	Impregnation	160/1.65–1.90, and 3.10	CO ₂ cycloaddition	Rate of up to 3.74 CO ₂ mmol min ⁻¹ ; increase of 733%	—	Reusability test, FT-IR, PXRD, SEM	—	176
Co(II)@TFPPy-PyTTA-COF	1,3,5-triformylphloroglucinol (TP), 2,5-diaminobenzenesulfonic acid (Pa-SO ₃ H)	PSM	600/1.2	CO ₂ cycloaddition	Yield: up to 96%	—	Reusability test, FT-IR, PXRD, SEM	—	177
p2NPh-OSO ₃ H HCP	Tetrakis(4-formylphenyl) pyrene (TFPPy) and pyrene-1,3,6,8-tetraol tetraamine (PYTTA)	PSM	180/~0.8	Using acetic anhydride to acylating phenols, alcohols, thiols, amines, and aldehydes	Conv.: 70–99%	—	Reusability test, FT-IR, PXRD	—	178
PAF-30-SO ₃ H	2-naphthol (NPh-OH)	PSM	174/2.8	Alkylation of guaiacol	Yield: 75–98%	1.14	Reusability test	—	178
PAF-20-SO ₃ H	Tetrakis(<i>p</i> -bromophenyl)methane and 4,4'-biphenyl(diboronic acid	PSM	354/2.3	Alkylation of guaiacol	Conv.: ~70% (yields: 34% mono and 35% dialkylate)	2.4	—	—	179
Pr-PAF-30-SO ₃ H	Tetrakis(<i>p</i> -bromophenyl)methane and 1,4-benzenediboronic acid	PSM	193 (PAF-30-SO ₃ H)/-	Alkylation of guaiacol	Conv.: 33% (yield: 23% C-isopropyl guaiacol e.g. from guaiacol: methoxycyclohexanol (32%), methoxycyclohexane (32%) and cyclohexane (27%))	1.4	—	—	131



Table 2 (continued)

POP	BBS/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
COF	1-Vinylimidazole (VIM), 2,5-dimethoxybenzene-1,4-dicarboxaldehyde (DMTPA), and 1,3,5-tri(4-aminophenyl)benzene (TAPB).	Imidazole alkylation with sultone	504/2.5	Biginelli reaction	Large scope, yields >90%	3.75% (Elemental analysis)	TGA, organic solvents, acids and reductants	Yes, aerogel-based cup-like reactor	132
COF	1,4-terephthalaldehyde and meso-tetrakis(4-aminophenyl)porphyrin	Amine reduction and alkylation with sultone	60/2-4	Dihydro-2-oxopyrrol four component synthesis	Yields > 85%	1.70	TGA and recovered catalyst analysis	No	180
Cashew nut shell liquid (CNSL)-formaldehyde resin	CSNL and formaldehyde	FC alkylation with 8-ClSO ₃ H	8-/	Bis(indolyl)methane and Di[bis(indolyl)methane] synthesis	Large scope, >80% yield	7.96-9.56% S (elemental analysis)	TGA	No	181
HCP	Bisphenol	FC alkylation with 452/0.55 ClSO ₃ H	452/0.55	Phosphonate synthesis	Large scope, yields > 90%	—	TGA	No	182
COF	Tri(4-formyl phenoxy) cyanurate (TFPC) and 1,4-diaminebenzene	FC alkylation with 12/0.12 ClSO ₃ H	12/0.12	Triazolo quinazolinones and 4H-pyrimido benzothiazoles synthesis	Large scope, yields 60-90%	30% weight loss (TGA)	TGA	No	183
COF	1,3,5-triformyl phloroglucinol (TFPG) and DABSA	<i>De novo</i>	251/-	3-component hydroxy carbamate synthesis	Yields 60-90%	11% (EDS)	No	No	184
CTF	Melamine	FC alkylation with — ClSO ₃ H	—	5-hydroxymethylfurfural (5-HMF) production	Temperature-time-catalyst ratio optimization	1.3	TGA	No	185
SBA-15 decorated CTF	Cyanuric chloride, melamine	Amine alkylation with sultone	157/2.1	5-HMF production	Production from maltose	5.5% weight loss of sulfonic groups (TGA)	TGA	No	157
HCP	Vinyl benzyl chloride, DVB	FC alkylation with 550-693/1 ClSO ₃ H	550-693/1	5-HMF production	Similar or better activity than <i>PTSA</i>	3.35-4.14	No	No	186
COF	TFPG and DABSA	<i>De novo</i>	159/1.4	5-HMF production	One pot synthesis of diformylfurfural from fructose	3.15	TGA	No	120
HCP	1,3,5-triphenylbenzene and 1,2-dichloroethane	H ₂ SO ₄ treated	234/3.91	5-HMF production	Solvent selection	2.73	TGA, recovered catalyst analysis	No	187
CTF	1,3-dicyanobenzene, 2,6-pyridinedicarbonitrile, 1,4-dicyanobenzene or 4,4'-biphenyldicarbonitrile	Gas phase sulfonation with SO ₃	600-2400/-	Hydrolysis of cellobiose	86% conv. 90% sel.	0.38-0.61	TGA	No	188
COF	Triaminophenylbenzene (TAPB) and 2,6-dimethoxyterephthalaldehyde (DMTF)	Ionic liquid encapsulation	1266-2231/2.2-2.7	Dehydrative formation of isosorbide from sorbitol	100% conv. and >91% select.	0.33-1.23	Basic treatment	No	189
PDVB	DVB and sodium <i>p</i> -styrenesulfonate hydrate (SPSS)	Ion exchange with dilute sulfuric acid	243-1008/4.3-9.49	Dehydrative formation of isosorbide from sorbitol	80% yield and 80% select.	0.29-3.73	Acid treatment, TGA	No	190
Hyper-crosslinked hollow spherical polymers	BHN (2,6-bis(hydroxymethyl)naphthalene) and BHB (1,4-bis(hydroxymethyl)benzene)	FC alkylation with ClSO ₃ H	323-365/2.74-6.31	FAME formation	Esterification of real samples with yield 64-92.3%	0.45-3.13	TGA	No	191
HCP	DCX, naphthalene	FC alkylation with ClSO ₃ H	486-816/-	Tetrahydropranylation of alcohols and transesterification of waste oil with methanol	90% yield of free fatty acids in 4 h	2.7-3.2	TGA	No	192

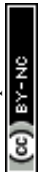


Table 2 (continued)

POP	BBS/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
HCP	4,4'-bis(chloromethyl)-1,1'-biphenyl, ClSO ₃ H	<i>De novo</i>	645–1093/-	Esterification of hexanoic acid with benzyl alcohol	95% conv. in 22 h run	1.26–4.44	TGA	No	193
HCP	Carbazole and α,α' -dibromo- <i>p</i> -xylene	FC alkylation with ClSO ₃ H	346/-	Biofuel synthesis from free fatty acids	Esterification with real samples in yields >90%	3.7	TGA	No	119
HCP	Phenol or bisphenol A	FC alkylation with ClSO ₃ H	210–324/7	Biofuel synthesis from free fatty acids	Esterification with real samples in yields 30–90%	0.85–1.60	TGA	No	163
HCP	1,3,5-triphenylbenzene and diphenylphosphate	FC alkylation with ClSO ₃ H	847/0.62–2.7	Esterification of levulinic acid	5 examples with C1–C5 alcohols in high conversion and selectivity	1.24	Stability after reuse	No	194
PDVB	DVB	FC alkylation with ClSO ₃ H	280–587/3.6–40.1	Esterifications and acylations	Conversion 47–88% with high selectivities >97.5%	2.13–3.72	TGA	No	118
COF	1,3,5-triformyl phloroglucinol, DABSA	Functionalized with 1-methyl-3-(3-sulfofropyl)-1 <i>H</i> -imidazol-3-ium hydrosulfate	103–264/7.4–8	Biodiesel formation	91% conversion of the model soybean oil into FAMES	Hammett function = 1.45–1.6	TGA and reused material analysis	No	195
HCP	1,3,5-tri(4-vinylphenyl)-benzene and 1,4-divinylbenzene	FC alkylation with ClSO ₃ H	478–738/4.2–8.7	Biodiesel formation: esterification of oleic acid with methanol	Yields > 95% with TOF: 44–100 mmol g ⁻¹ h ⁻¹	3.1–3.6	TGA	No	116
PAF	1,3,5- <i>tris</i> (4-Bromophenyl) benzene, benzene-1,4-diboronic acid and 1,3,5- <i>tris</i> (4-bromophenyl)amine	Treatment with oleum	210–270	Esterification of acetic acid with <i>n</i> -butanol	Conversion > 70%, superior to benchmark Amberlist 15	5	TGA	No	122
Hollow porous organic polymer	Tetrakis(4-ethynylphenyl) methane and 1,4-diodobenzene	FC alkylation with ClSO ₃ H	232/<2	Polyketone synthesis	~ 5 g of polymer per 1.5 mg of catalyst without fouling the reactor	6.91% S (Elemental analysis)	NMR	Yes	121
Silica - porous organic polymer	Tetrakis(4-ethynylphenyl) methane and 1,4-diodobenzene	FC alkylation with ClSO ₃ H	41–90/1–3	Terpolymerization of CO ₂ , ethylene, and propylene	17.2 kg of polymer per gram without fouling the reactor	0.28–0.47	DSC	Yes	196
PDVB	DVB and 4-vinylbenzenesulfonic acid sodium salt	<i>De novo</i> monomer truncation	228–542/1–10	Polypropylene synthesis	36.4 kg of polymer per gram with high stereoregularity	1.43	TGA - DSC	Yes	197
PDVB	DVB and 4-vinylbenzenesulfonic acid sodium salt or sodium allylsulfonate	<i>De novo</i> monomer truncation	385–471/-	Polyethylene synthesis	7 g of polymer per g of catalyst per h with ultrahigh molecular weight	—	No	No	198
COF	4,4'-biacetyl-2-hydroxyphenyl	–OH functionalization with sultone	362/2	Cascade synthesis of benzimidazoles, benzodiazepines and reductive amination of carbonyls	Large scope, yields > 90%	2.87	TGA	No	199
HCP	Tetraphenylethylene and formaldehyde dimethylacetal	FC alkylation with ClSO ₃ H	995/1–4	Synthesis of symmetrical triarylmethanes	Large scope, different nucleophiles, yields: 30–95%	11.11	TGA and reused material characterization	No	200
PDVB	DVB and 4-styrenesulfonic acid sodium salt	<i>De novo</i> monomer truncation	250–535/4.5–15.7	Deacetalization – Knoevenagel condensation	100% conversion 99% yield of the cascade product	0.58–3.5	TGA	No	162



Table 2 (continued)

POP	BBS/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
CTF	Cyanuric chloride and <i>p</i> -terphenyl	FC alkylation with ClSO ₃ H	223/0.6–5.5	One pot hydrogenation-esterification	90% yield to cascade product in 8 h, reusable material	2.99	No	No	201
COF	1,3,5-triformyl phloroglucinol, DABSA	Functionalized with (1 <i>R</i> ,2 <i>R</i>)-1,2-diaminocyclohexane	45/1.5	Asymmetric Michael addition	Enantiomeric excess: 84 – 96%	1.65	NMR	No	202

and further modification seem to be valid strategies for the development of efficient catalysts in organic transformations. As a whole, different types of polymeric structures may be suitable to be employed in organic reactions, and the control on the functionalization results in solid acids that present advantages in important organic reactions. However, the choice of one platform over another remains an unresolved question, and current mechanistic understanding is limited by the lack of precise characterizations, especially regarding the nature of active sites and in-operando measurements. Indeed, vague and simplistic homogeneous mimics are commonly found. These aspects demand further and deeper investigation that integrates theory and unconventional characterization.

A key advantage of SPOPs is their tunable multifunctionality, which is critical for upgrading complex, real-world feedstocks. Wang *et al.* engineered a bifunctional acid–base HCP (HCP-BBA-*S*-*x*) for the *in situ* catalytic pyrolysis of oil shale (Fig. 4a).¹⁷³ This system exemplifies the structure–performance trade-offs inherent to SPOP design and their superiority over single-function catalysts. The study revealed a non-linear relationship between acid-site density and performance. While increased sulfonation (variable “*x*”) raised acid density from ~0.9 to ~3.0 mmol g⁻¹, excessive functionalization (*x* > 0.5) caused pore blockage, collapsing the BET surface area from ~1050 m² g⁻¹ to ~350 m² g⁻¹ and diminishing activity. The optimal catalyst, HCP-BBA-*S*-0.5, balanced a moderate acid density (~2.7 mmol g⁻¹) with preserved hierarchical porosity (avg. pore size ~4.5 nm), achieving a 19% increase in shale oil yield (~42%) over the non-catalytic process. Mechanistically, the proximity of acidic (–SO₃H) and basic (–NH₂) sites within the rigid framework enabled synergistic pathways: acid sites cracked heavy hydrocarbons, while adjacent base sites facilitated deoxygenation and isomerization. This cooperation directly enhanced fuel quality, reducing oxygen content by 55% (to 1.64%) and increasing the H/C ratio to 1.80. This case provides a critical design principle for SPOPs in demanding applications, an optimal balance between acid-site density, pore integrity, and site cooperation is more crucial than maximizing any single property. This tunable, multifunctional design surpasses the capabilities of conventional single-site solid acids.

In a complementary study focusing on controlled acid-site incorporation, Wei and co-workers demonstrated an alternative synthetic approach *via in situ* sulfonation during polymerization. Their catalyst, PDVB-SO₃H-IS, was prepared by copolymerizing sodium *p*-styrenesulfonate (also referred to in the literature as sodium vinylbenzenesulfonate or SPSS) with DVB, followed by proton exchange, yielding a mesoporous polydivinylbenzene-based solid acid with an acid density of 1.58 mmol H⁺/g.¹⁷⁴ This material exhibited excellent performance in the acid-catalyzed condensation of phenol and acetone to produce bisphenol A, achieving ~38% phenol conversion (near the theoretical limit) and >90% selectivity at 90 °C (Scheme 3). Its robust reusability over five cycles was attributed to the covalent anchoring of –SO₃H groups within the cross-linked polymeric matrix, preventing leaching.

A pivotal study by Woodward *et al.* redefined the synthesis of SHCPs through a one-pot, metal-free methodology using ClSO₃H as both a FC catalyst and sulfonating agent.¹⁵⁴ This

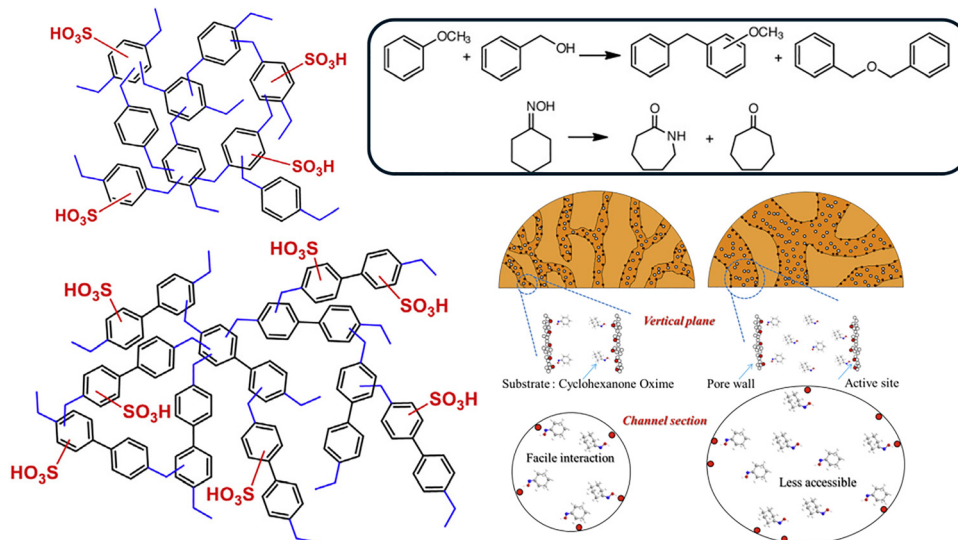


Fig. 3 HCPs with activity in FC alkylation and Beckmann rearrangement, and a plausible explanation. Reproduced with permission from ref. 168 Copyright 2015 Elsevier.

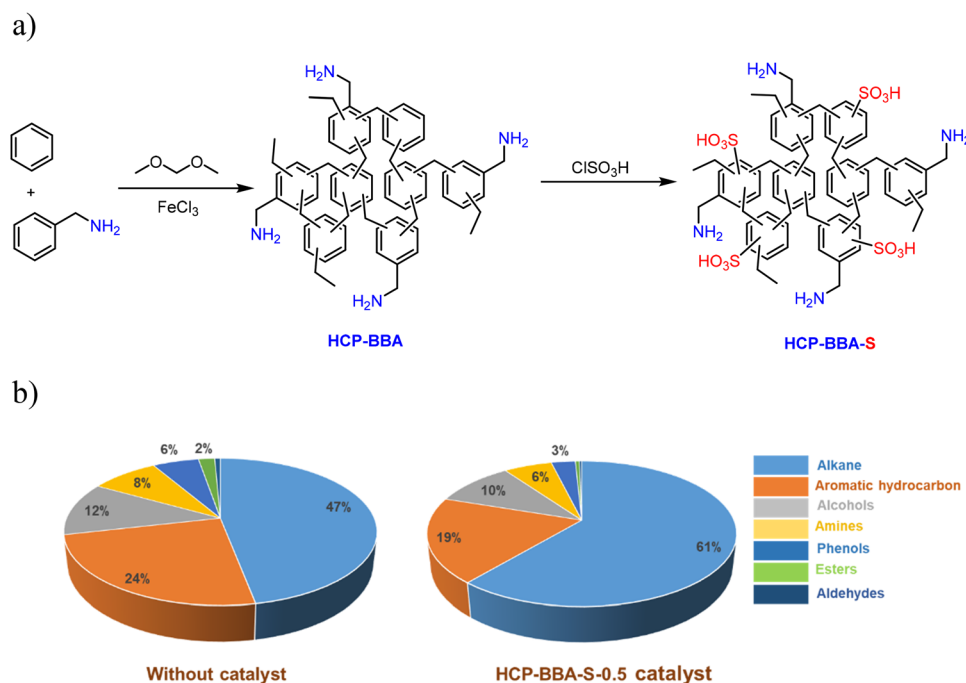
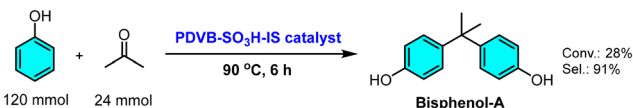


Fig. 4 Catalyst synthesis and its performance: a) preparation steps for the HCP-BBA-S-x catalysts; b) comparative composition of shale oil produced with and without the HCP-BBA-S-0.5 catalyst. Adapted from ref. 173.

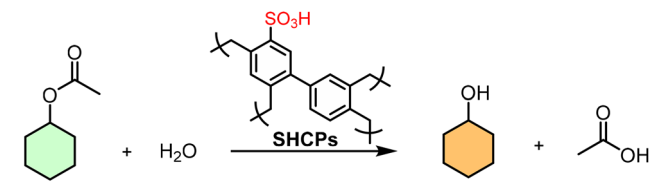


Scheme 3 Phenol conversion and bisphenol-A selectivity achieved using PDVB-SO₃H-IS as a catalyst. Adapted from ref. 174.

integrated approach condensed a conventionally multi-step, multi-day process into under 24 h, eliminating separate metal catalysts and significantly reducing reagent use. By simply

varying the ClSO₃H-to-monomer molar ratio, a series of SHCPs (SHCP-1 to SHCP-10) with finely tuned acid densities (up to ~3.8 mmol g⁻¹) could be prepared. They found that SHCP-5, with an optimal acid density of 2.20 mmol g⁻¹ and a preserved BET surface area of 890 m² g⁻¹, achieved ~75% conversion in hydrolyzing cyclohexyl acetate, surpassing both Amberlyst-15 and homogeneous H₂SO₄. This study underscores that maximum -SO₃H loading can compromise porosity; optimal performance arises from a synergistic interplay of both properties (Table 3).



Table 3 Conversion and selectivity of SHCPs in the catalytic hydrolysis of cyclohexyl acetate to cyclohexanol¹⁵⁴


Catalyst	-SO ₃ H content (mmol g ⁻¹)	BET surface area (m ² g ⁻¹)	Conversion ^a (mol%) (with selectivity > 99%)
SHCP-1	0.03	~540	0
SHCP-2	0.44	~900	~31
SHCP-3	1.15	~1060	~68
SHCP-4	1.72	~950	~72
SHCP-5	2.20	~890	~75
SHCP-6	2.82	~805	~72
SHCP-7	3.24	~725	~64
SHCP-8	3.40	~705	~68
SHCP-9	3.60	~680	~65
SHCP-10	3.76	~690	~59

^a Calculated by GC-MS.

The strategic integration of amphiphilicity into SPOPs presents a powerful approach to overcome mass-transfer limitations in aqueous-phase catalysis. In a study, an amphiphilic sulfonated HCP (SHCP) catalyst series, denoted P(QP-SBA)-*n* (with *n* being the SBS-to-QP molar ratio), was synthesized *via* solvothermal copolymerization of a phosphonium-based surfactant monomer (QP) and vinyl benzenesulfonate (SBS, previously referred to as SPSS), followed by acidification.¹²⁹ The design rationale centered on creating a polymer matrix with intrinsic swelling behavior and dual hydrophilic-hydrophobic character. The optimal catalyst, P(QP-SBA)-3, proved highly effective for hydrolyzing cyclohexyl acetate (93% conversion, 95% selectivity, 89% yield), outperforming conventional acids (H₂SO₄, PhSO₃H) and benchmark solid acids like Amberlyst-15, [HSO₃b-mim]HSO₄ IL,²⁰⁶ and high-surface-area carbon solid acid.²⁰⁷ Critical analysis of the structure-performance relationship reveals that its superiority is not merely a function of high acid density (~0.36 mmol g⁻¹) but stems from a synergistic trifecta: (i) favorable BET surface area (228 m² g⁻¹) enabling accessible sites, (ii) excellent surface amphiphilicity (evidenced by near-zero contact angles for both water and organic phases) ensuring effective dispersion and interfacial contact, and (iii) a swollen gel-like phase that enriches hydrophobic substrates within the polymer network, effectively mimicking phase-transfer catalysis. This holistic design enabled efficient catalysis even under static conditions, highlighting reduced energy input for mixing. The catalyst maintained robust performance over four cycles, with minor deactivation attributed primarily to physical handling losses, underscoring the structural stability imparted by the framework.

Moving beyond traditional acid catalysis, the functionality of -SO₃H groups can be leveraged to modulate the electronic structure and coordination environments within sophisticated frameworks for advanced applications. A seminal example is the work by Li *et al.*, who strategically incorporated -SO₃H groups

into the organic struts during the synthesis of a β-ketoenamine COF.¹⁷⁵ This design served a dual purpose: first, the -SO₃H groups induced a morphological transformation from nanorods to interwoven nanowires, significantly increasing the specific surface area from 320 to 600 m² g⁻¹ and creating a more favorable environment for metal anchoring. Second, and more critically, these electron-withdrawing groups acted as electronic modulators for the subsequent immobilization of single-atom cobalt sites. Through comprehensive characterization including XPS, X-ray absorption fine structure (XAFS), high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) and density functional theory (DFT) calculations, the study elucidated that the sulfonic functionality cooperatively tuned the local Co-N-C coordination (Fig. 5).

This modulation enhanced the Lewis acidity of the Co center while the proximal sulfonic groups provided Brønsted acid sites, thereby facilitating proton transfer. Consequently, the engineered solid acid catalyst, S-Co/NC, demonstrated exceptional performance for the energy-intensive solvent regeneration step in amine-based CO₂ capture. By accelerating carbamate decomposition *via* synergistic Lewis and Brønsted acid catalysis, S-Co/NC increased the reaction rate by over 700% relative to the non-catalytic baseline. This enhancement facilitated effective CO₂ desorption at a reduced temperature, which corresponded to an approximately 29% decrease in the energy required for solvent regeneration. This study transcends conventional use of -SO₃H as merely a Brønsted acid site; it showcases its pivotal role as a morphological director and electronic modulator in constructing advanced single-atom catalysts, opening new avenues for SPOPs in addressing critical energy and environmental challenges.

Moving beyond simple impregnation, a sophisticated host-guest design strategy leveraging acid-base interactions was employed to construct a highly selective heterogeneous catalyst. Xue *et al.* ingeniously immobilized an alkaline amino acid ionic liquid (AIL), [DBUH]₂Cys (synthesized from 1,8-diazabicyclo[5.4.0]undec-7-ene and L-cysteine), within the precisely defined nanochannels of a -SO₃H-functionalized COF (TpPa-SO₃H).¹⁷⁶ The resulting composites, [DBUH]₂Cys@COF-*X* (*X* = 1-4; 0.1-0.4 mmol IL), were applied to CO₂ cycloaddition reaction (Fig. 6). This design achieved two key outcomes: (1) it created confined bifunctional active sites within the pores, and (2) it engineered the pore environment. While IL incorporation reduced the BET surface area, it strategically increased pore size and volume, creating defined mesopores (~3.0 nm) in the optimal catalyst, [DBUH]₂Cys@COF-3. This engineered nanostructure translated directly to exceptional and selective performance. The catalyst achieved a 96% yield for the epichlorohydrin conversion under remarkably mild, solvent-free conditions (1 atm CO₂, 100 °C). More importantly, it exhibited pronounced size-selective catalysis: small epoxides (*e.g.*, epichlorohydrin, glycidol) were converted near-quantitatively (95-96%), while bulky substrate (*e.g.*, styrene oxide) was sterically excluded, yielding only trace amounts. This transforms the composite into a molecular gatekeeper, a rare property for solid acids. Consequently, this system outperforms many reported IL/COF hybrids,²⁰⁸⁻²¹¹ which often require higher pressure, loading, or co-catalysts. DFT calculations revealed



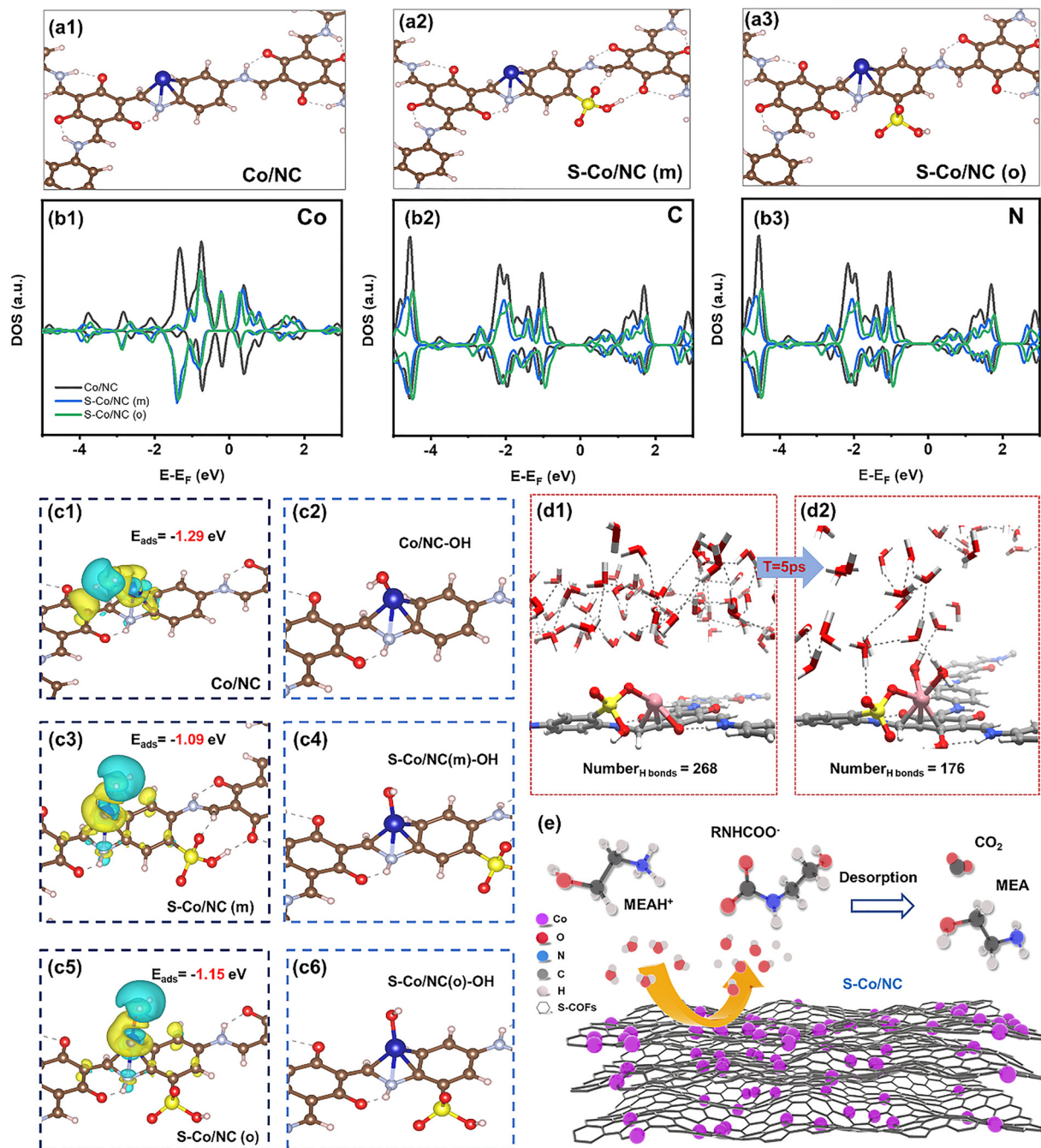


Fig. 5 Structural and electronic characterization of COF catalysts and the catalytic mechanism for carbamate decomposition. a) Atomic structure of the Co active sites in (a1) Co/NC and (a2) S-Co/NC, with their corresponding reaction pathways; b) Electronic structure analysis showing the density of states for (b1) Co, (b2) C, and (b3) N atoms in different catalysts; c) Charge redistribution upon H₂O adsorption (blue/yellow) and optimized catalyst structures with surface –OH groups; d) *Ab initio* molecular dynamics simulation of H₂O interaction with the S-Co/NC surface; e) Proposed catalytic cycle for carbamate decomposition over the S-Co/NC catalyst. Monoethanolamine (MEA) was used as the benchmark CO₂ capture solvent. Reproduced with permission from ref. 175 Copyright 2024 Elsevier.

the synergistic mechanism enabled by confinement: the [DBUH]⁺ cation activates the epoxide *via* H-bonding, while the proximal cysteine carboxylate (–COO[–]) activates CO₂ and attacks the epoxide, cooperatively lowering the energy barrier. While the catalyst was recyclable over three cycles, activity loss in the fourth cycle highlighted an ongoing challenge with IL leaching stability. Nevertheless, this work exemplifies how precise spatial

organization of functional groups within SPOPs can unlock unique selectivity and cooperative catalysis under mild conditions.

An ionic, heterogeneous 2D COF, denoted as Co(n)@TFPPy-PyTTA-COF, was synthesized *via* a multi-step PSM of the imine-linked TFPPy-PyTTA-COF (where TFPPy = tetrakis(4-formylphenyl)pyrene and PyTTA = pyrene-1,3,6,8-tetrayl tetraaniline) (Scheme 4).¹⁷⁷ The PSM process involved: (i) reduction of the



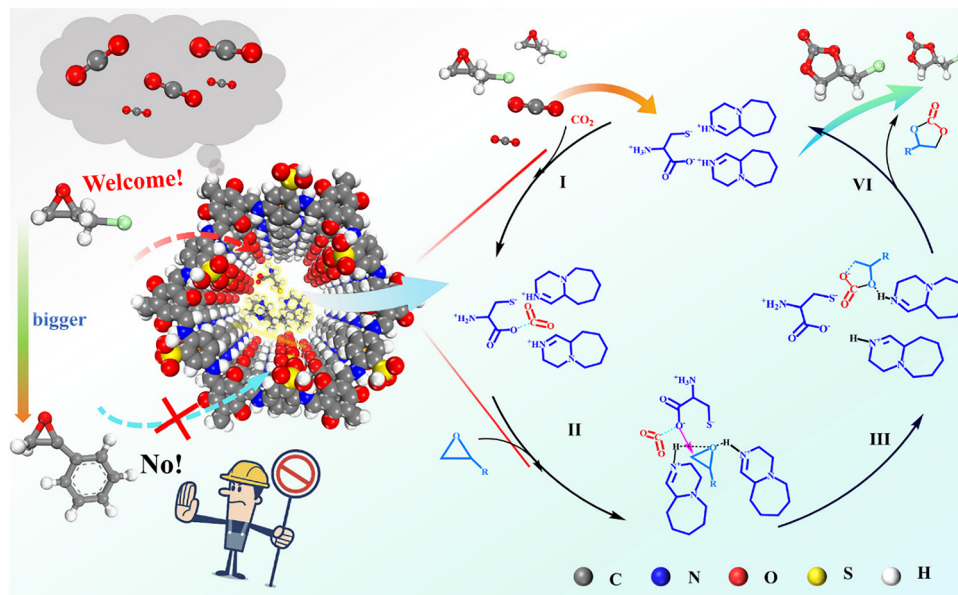
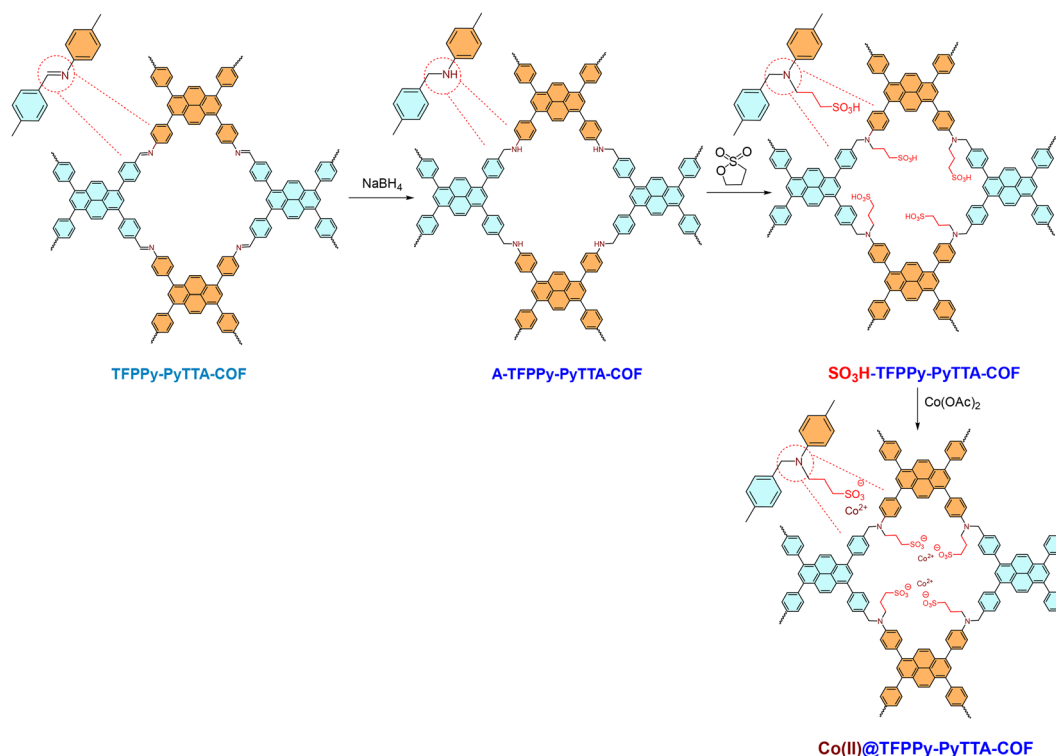


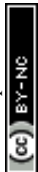
Fig. 6 Proposed catalytic mechanism for the synergistic cycloaddition of CO_2 with epoxides using $[\text{DBUH}]_2\text{Cys}@\text{COF}-3$ as a catalyst. Reproduced with permission from ref. 176 Copyright 2024 Elsevier.



Scheme 4 Schematic illustration of the PSM process for synthesizing $\text{Co(II)}@\text{TFPPy-PyTTA-COF}$. Adapted from ref. 177.

imine ($\text{C}=\text{N}$) linkages to amine ($\text{C}-\text{N}$) groups, (ii) ring-opening alkylation of these amines with 1,3-propane sultone to install alkyl sulfonate groups, and (iii) metalation with cobalt(II) acetate to anchor Co(II) ions *via* ionic coordination to the sulfonate moieties, yielding $\text{Co(II)}@\text{TFPPy-PyTTA-COF}$. This linkage-focused PSM is significant because it bypasses incompatibility

between functional groups and harsh solvothermal synthesis conditions, enabling precise installation of active sites within pre-formed, stable pores. The resulting ionic, bifunctional pore environment featuring Lewis acidic Co^{2+} centers and nucleophilic sulfonate anions proved highly effective for the converting CO_2 and epoxides into cyclic carbonates. Under the



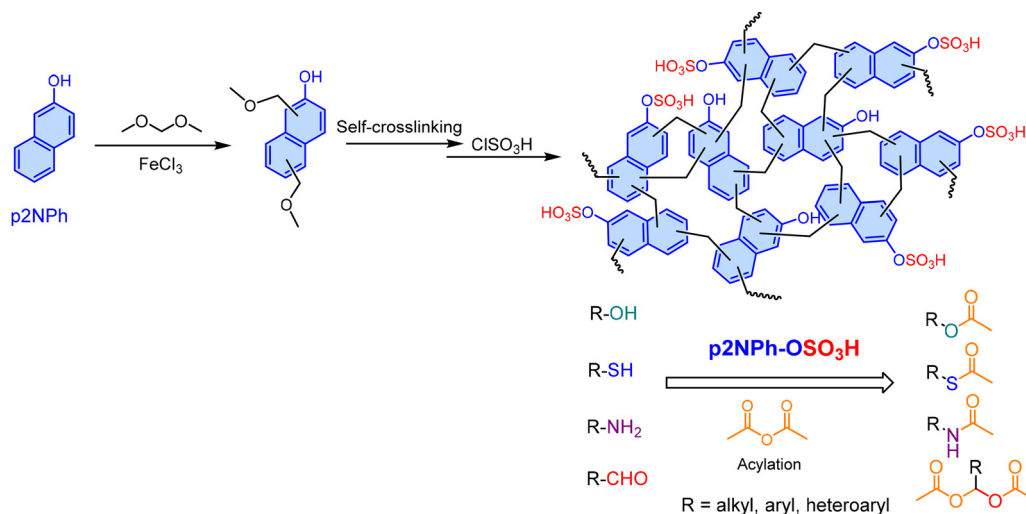
optimized reaction parameters (100 °C, 3.0 MPa CO₂), the catalyst achieved high conversions (70–99%) and yields (65–99%) for a range of epoxides, matching the activity of the homogeneous Co(OAc)₂. More importantly, it exhibited heterogeneity and stability, retaining its crystallinity and activity over five catalytic cycles without metal leaching. This work establishes linkage engineering *via* PSM as a powerful strategy for integrating cooperative acid–base sites into robust porous scaffolds, creating high-performance, reusable catalysts for efficient CO₂ utilization.

Kim and colleagues¹⁷⁸ synthesized a 2-naphthol-based HCP (p2NPh-OH) *via* FC alkylation using dimethoxymethane as the cross-linker (Scheme 5). Subsequent functionalization with ClSO₃H yielded the sulfonated solid acid catalyst, p2NPh-OSO₃H. This recoverable catalyst exhibited excellent performance for the protection and acylation of diverse substrates including phenols, alcohols, thiols, amines, and aldehydes using acetic anhydride under solvent-free conditions at room temperature. In the acylation of benzyl alcohol, p2NPh-OSO₃H demonstrated superior catalytic activity compared to both polydopamine-derived sulfamic acid-functionalized magnetic nanoparticles (Fe₃O₄@PDA-SO₃H),²¹² noting its tedious synthesis, and Ph₃P(OAc)₂.²¹³ Notably, the catalyst exhibited no significant loss in activity after ten cycles, confirming its high stability and reusability.

The well-defined nanoporosity of SPOPs acts as a nanoreactor, imposing spatial constraints that can dramatically influence reaction pathways and selectivity, moving SPOPs beyond being mere acid reservoirs to becoming precision tools for selective synthesis. This is exemplified by the alkylation of guaiacol over two sulfonated PAF, PAF-20-SO₃H and PAF-30-SO₃H.¹⁷⁹ Their synthesis *via* Suzuki cross-coupling and post-sulfonation created a deliberate structural contrast: PAF-20-SO₃H had higher surface area but smaller pores (~2.3 nm), while PAF-30-SO₃H featured larger pores (~2.8 nm) that facilitated a higher –SO₃H loading (2.34 vs. 1.88 mmol g^{−1}) (Scheme 6). This variance directly dictated catalytic selectivity. With isopropanol, the larger-pore

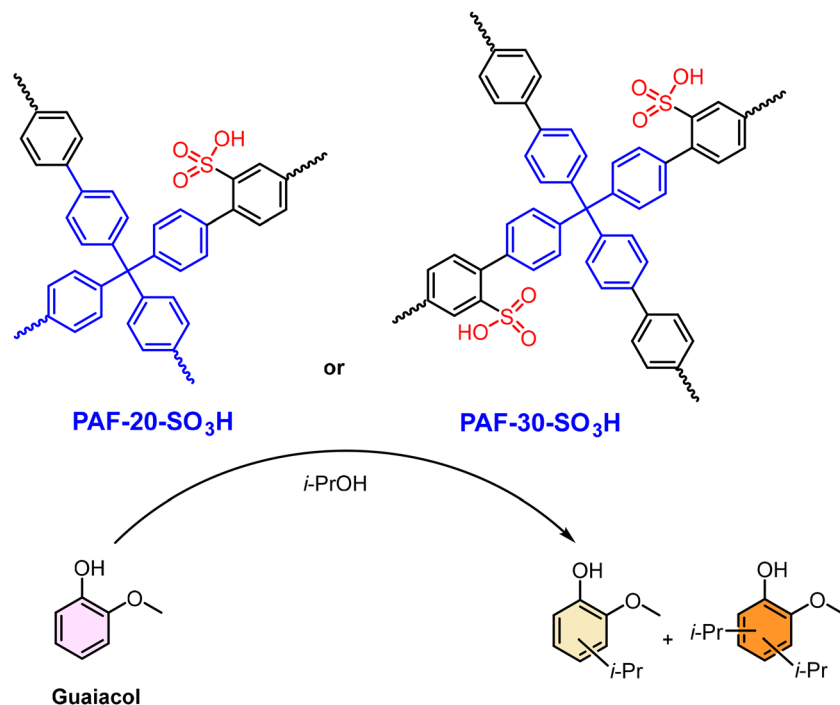
PAF-30-SO₃H achieved higher conversion (~70% vs. ~30% in 6 h) and yielded both mono- and dialkylates. In contrast, the more confined pores of PAF-20-SO₃H favored monoalkylation, suggesting steric control over product distribution. The challenge of reactant competition was evident with bulkier cyclohexanol, where its dehydration product (cyclohexene) blocked sites. However, process optimization using a non-polar solvent and limiting alcohol concentration enabled PAF-30-SO₃H to achieve ~60% guaiacol conversion to valuable cyclohexyl derivatives in 2 h. This case demonstrates that for SPOPs in reaction systems with competing chemical pathways, achieving high performance requires a dual-design approach: engineering the pore nanoconfinement to control selectivity and simultaneously optimizing the bulk reaction environment to manage these pathways.

In related work, Kalinina *et al.* directly compared a platinum catalyst supported on a non-acidic PAF (Pt-PAF-30) with its sulfonated analog (Pt-PAF-30-SO₃H) for the hydrodeoxygenation (HDO) of bio-derived molecules (guaiacol, veratrole, catechol) (Scheme 7).¹³¹ The contrast in catalytic pathways was stark. Pt/PAF-30 primarily hydrogenated aromatic rings (*e.g.*, converting guaiacol to methoxycyclohexanol), with deoxygenation limited to partially hydrogenated intermediates. Critically, it could not perform demethoxylation on saturated rings. In contrast, Pt-PAF-30-SO₃H efficiently catalyzed both dehydroxylation and demethoxylation of saturated species, dramatically boosting the yield of fully deoxygenated cyclohexane to 65–95%. This underscores that proximal acid sites are essential for cleaving robust C(alkyl)–O bonds (*e.g.*, in methoxycyclohexane), a step often rate-limiting in HDO. The synergy likely involves acid-catalyzed dehydration or methanol elimination from protonated intermediates, a pathway inaccessible to the metal site alone. This work exemplifies how integrating –SO₃H functionality transforms a simple hydrogenation catalyst into a highly effective deoxygenation system, a design principle leveraged in subsequent work for broader lignin model substrates.²¹⁴

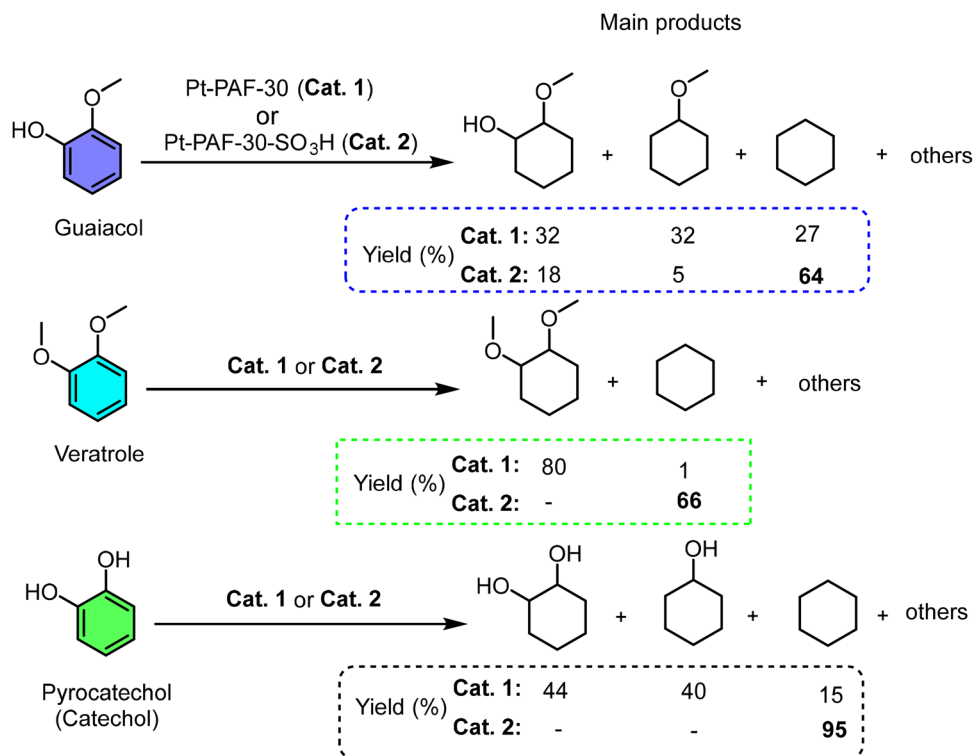


Scheme 5 Synthetic route to sulfonated poly(2-naphthol) and its use as a catalyst for the acylation of various functional groups with acetic anhydride. Adapted from ref. 178.





Scheme 6 Reaction scheme for the alkylation of guaiacol with isopropanol catalyzed by PAF-20-SO₃H and PAF-30-SO₃H, showing the main products. Adapted from ref. 179.



Scheme 7 PAF-SO₃H catalyzed hydrogenation-deoxygenation of lignin-derived model compounds. Adapted from ref. 131.

3.2. Heterocycle and pharmacologically relevant compounds

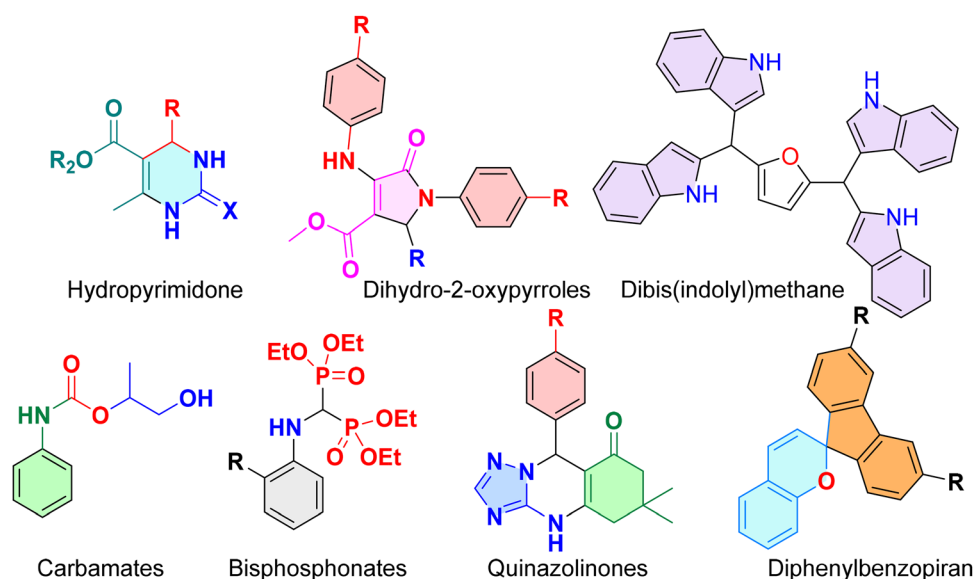
The synthesis of N- and O-heterocycles, crucial pharmacophores, often requires precise acid catalysis and efficient

product separation.^{215–218} SPOPs offer a unique platform for these transformations by combining strong and accessible Brønsted acid sites with a tunable hydrophobic/hydrophilic



pore environment, which can influence substrate preorganization and selectivity. In the previous section, Yevad *et al.*, have discussed the synthesis of diphenylbenzopirans with good activity.¹⁷² This work represents a good example of the power of SPOPs for the synthesis of relevant heterocyclic compounds (Scheme 8). In this sense, Yao and coworkers presented a COF created in a multicomponent Povarov reaction among polytopic aldehydes and amines with vinyl-imidazole.¹³² Hence, they made a porous material connected by strong quinoline linkages compared to the standard and labile imine bonds to increase robustness. In addition, the pending imidazole allowed an easy PSM to introduce the $-\text{SO}_3\text{H}$ by a straightforward alkylation with 1,3-propanesultone. As a result, a $-\text{SO}_3\text{H}$ -ionic liquid COF was yielded. The combination of both moieties was concluded responsible for the activity in the catalytic Biginelli hydroxypyrimidone formation reaction studied: yields above 90% with excellent recyclability and even a scale-up attempt at 0.3 mmol level were reported. Authors took advantage of heterogeneous catalyst properties in terms of activity, robustness and straightforward separation to outperform typical homogeneous catalysts (*e.g.* PTSA) and facing a greener effort in the Biginelli reaction. A similar synthetic route was followed by our group.¹⁸⁰ A porphyrin-iminic COF was reduced and furtherly alkylated with 1,3-propanesultone. This facile $-\text{SO}_3\text{H}$ functionalization underscores the synthetic versatility of the polymeric platform, enabling the efficient synthesis of highly substituted dihydro-2-oxypyrrroles in high yield (>90%) within 60 min. A comparative study with typical catalysts for that transformation was included, and the resulting polymer presented better results than homogeneous (*e.g.* I_2)²¹⁹ and heterogeneous systems (*e.g.* Fe_3O_4 @nanocarbon) catalysts.²²⁰ This enhanced performance is attributed to a synergistic effect arising from two key features: the strong Brønsted acidity of the anchored $-\text{SO}_3\text{H}$ groups and the porous structure of the polymer framework. Kotkar and Tilve presented an acidic resin from cashew nutshell liquor, a biomass waste product, for

synthesizing bis(indolyl)methanes from xylochemicals.¹⁸¹ This work addresses a significant gap, as such products are rarely synthesized using catalysts derived from biomass, let alone SPOPs from this source. To create the catalyst, they performed a stepwise oxidation of the liquor followed by sulfonation with ClSO_3H , yielding a low-cost polymer randomly decorated with carboxylic and $-\text{SO}_3\text{H}$ groups. Although the material had a non-uniform acid site distribution, its high total acidity (~ 2.8 mmol H^+/g) was proved highly effective. In the proposed mechanism, this acidity efficiently activates the aldehyde substrate for nucleophilic attack by indole. The SPOP's merits were demonstrated quantitatively: it achieved excellent yields (>90%) under mild conditions, outperforming conventional acid catalysts like silica-supported sulfuric acid. Crucially, the biomass-derived SPOP catalyst could be reused for multiple cycles with minimal activity loss, highlighting its robustness and potential economic advantage. This case underscores a key strength of SPOPs: their synthesis can be tailored using sustainable, low-cost feedstocks to create competitive, reusable solid acids, offering a simpler and safer alternative to traditional benchmark systems. *N*-Containing bisphosphonates could be efficiently obtained using a $-\text{SO}_3\text{H}$ functionalized HCP, according to Reddy *et al.*¹⁸² The decorated polymer was synthesized using FC polymerization of bisphenol crosslinked with dimethoxymethane, followed by the typical post-synthetic functionalization of HCPs consisting in the sulfonation. As a result, it produced a catalyst active in the three-component reaction of 2-(*tert*-butyl)aniline, triethyl orthoformate, and diethyl phosphite, affording the target product under solvent-free conditions at 50 °C in only 1 h, with excellent recyclability. Authors proposed a catalytic mechanism involving the presence of surface sulfonate groups as generic proton donors over suitable substrates as main reasons for their results, in what they considered the first example of HCP catalyzing that reaction. The easy synthesis, where the $-\text{SO}_3\text{H}$ resembles present and accessible for the construction of solid acids may be the key for achieving



Scheme 8 Relevant heterocyclic molecules prepared by sulfonated polymers.



this active material. In summary, a common mechanistic theme across these studies is the proposed role of sulfonate groups in activating reaction substrates *via* hydrogen bonding. However, the contribution of polymer backbone rigidity (COF *vs.* HCP) to stabilizing the resulting charged intermediates remains an area requiring deeper investigation, potentially through DFT studies and/or advanced spectroscopic/electrochemical techniques.

Zolfigol and coworkers showed that the incorporation of magnetic nanoparticles in the COF facilitated catalyst recovery without compromising $-\text{SO}_3\text{H}$ site accessibility, leading to superior performance in triazoloquinazolinone synthesis. This exemplifies the design principle of adding separable functionality orthogonal to the catalytic site.¹⁸³ Using only 20 mg of material, the target product was obtained with superior performance compared to benchmark catalysts for the same reaction (10 min, 80 °C, neat reaction, 93% yield). In this case, the motivation for having magnetic easy-to-separate material with acid sites becomes crucial, representing a very good example of efficient combination of the material properties. Mondal and coworkers employed an imine COF material where the sulfone was present in one of the monomers submitted to the polymerization to study the synthesis of alkylhydroxy carbamates using amines, epoxides and CO_2 in just 8 h of reaction.¹⁸⁴ This example shows a topological design and introduction of the $-\text{SO}_3\text{H}$, which may be present in all the material pores, in contrast with the other examples discussed in this section. As a result of that intrinsic acidity, a variety of products could be obtained. In addition, the material was robust enough to present good cyclability over five consecutive reaction runs, although pore blockage was observed after the 6th cycle, that hinders the full potential of this type of topologically designed COFs.

As demonstrated, sulfonated polymers can be used to construct relevant and complex architectures, and they can be achieved with different materials, cores and functionalization methods, although topological sulfonated materials presented pore blockage problems. Nevertheless, the acidity and accessibility of active sites in all materials decorated with $-\text{SO}_3\text{H}$ groups, combined with the facile recovery, was beneficial for the reaction and allowed the outperforming of benchmark and/or homogeneous catalysts. A major future challenge lies in quantitatively correlating acid strength distribution within SPOP pores with reaction kinetics and selectivity, moving beyond yield-based comparisons. Thus, a direct relationship between the different material types will shed light on these questions too.

3.3. Biomass valorization

Sulfonated polymers are also pivotal as acid catalysts in biomass valorization. Their strong acidic character makes them particularly well-suited for reactions involved in biomass valorization, a process that adds value to plant matter and organic residues. Chemically (Fig. 7), this often entails acid-catalyzed dehydration of glycosidic polymers such as cellulose, lignin, or biosugars, leading to the formation of furfural derivatives, which serve as valuable precursors for plastics and textiles, where the specific surface acidity and heterogeneous character of SPOPs may play a crucial role. Nevertheless, other processes where biomolecules

are transformed into commodities by acid catalysts are also considered under this definition. This section covers the role of SPOPs as acidic platforms for performing this important task framed in the greener effort towards finding a sustainable alternative for petrochemicals in the obtaining of highly functionalized useful chemicals.

Indeed, when sulfonated polymers are considered, 5-hydroxymethylfurfural (5-HMF) is one of the most fashionable products (Fig. 7).²²¹ Very recently, the work by Darvishi and coworkers demonstrates that a CTF functionalized with groups in a post-synthetic reaction is able to dehydrate fructose to 5-HMF with high efficiency (Fig. 7a).¹⁸⁵ CTF are indeed robust platforms that can host high loadings of sulfonation. Thus, they achieved 97% yield in this reaction (110 °C, 40 min, DMSO as solvent) using a CTF with 1.33 mmol of sulfonic acid per gram of material, with good recyclability due to the heterogeneous character and without leaching of active species, while the pristine material showed negligible activity. Babaei *et al.* achieved similar results with SBA-15 silica particles coated with a CTF layer that was finally functionalized on the amine that links the triazine rings instead of using the aromatic core (Fig. 7b).¹⁵⁷ In this study, the production of 5-HMF was first modeled using fructose, followed by the use of larger biomolecules such as sucrose, maltose, and glucose to provide insight into potential real-world applications. It seems that the employment of the robust core of CTFs can be beneficial for this particular acid-mediated dehydration. In addition, the accessibility and richness of aromatic cores allow an easy material functionalization with sulfonic acids prompted to trigger reactivity.

Nevertheless, other materials can be employed too. In another work, Du *et al.* compared the catalytic activity of a HCP functionalized on the aromatic core with $-\text{SO}_3\text{H}$ (sample S1) with a PAF functionalized in identical manner (sample S2) in the production of 5-HMF from fructose (Fig. 7c).¹⁸⁶ They observed that both materials matched the activity of PTSA in a reaction performed under microwave irradiation at 140 °C in a 4:1 mixture dioxane: water, and the activity was superior to Amberlist-15 catalyst. They concluded that the large and accessible surface area of S1 and S2 samples (550–693 $\text{m}^2 \text{g}^{-1}$) was responsible for the observed phenomena. An iminic COF also showed promising activity in the synthesis of 5-HMF.¹²⁰ The selected monomers contained the sulfonic moiety, and the material could obtain the target molecule in 1 h in DMSO with good conversion and selectivity. In addition, the acidic and reactive character of this sulfonated COF allowed authors to reach diformylfuran in moderate performance. Ye and coworkers studied the dehydration of fructose using a functionalized HCP in different solvents.¹⁸⁷ They obtained the best conversion of fructose and selectivity towards 5-HMF using DMSO (100% and 85%, respectively) among DMF, THF or water, but it deactivated quickly. Thus, they switched to a mixture of dioxane-water, and their study pointed out that a small amount of water in the mixture (2.5% v/v) could depress the oligomerization of 5-HMF and enhance the stability of the material. Common trends are therefore observed, indicating that a range of SPOP architectures can achieve high activity (*e.g.*, yields up to 97% under optimal conditions) in 5-HMF synthesis, often matching or surpassing



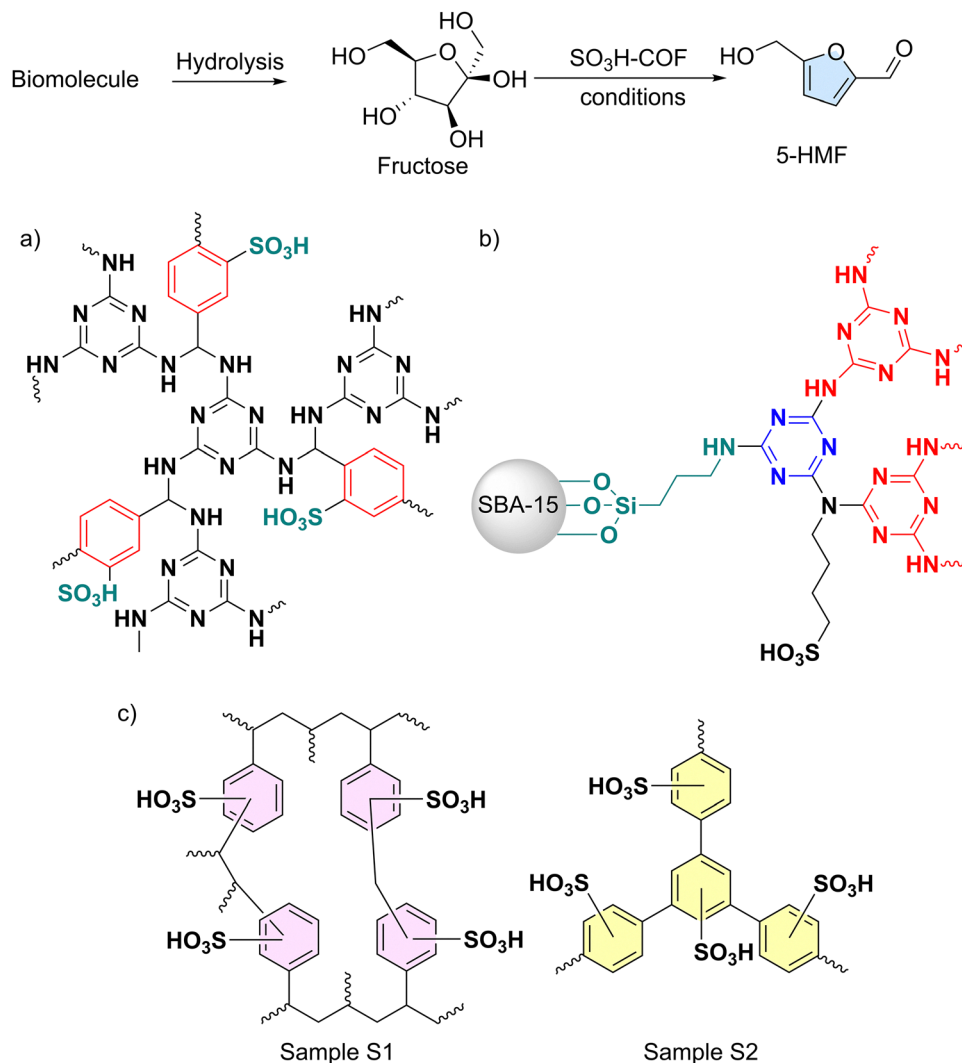


Fig. 7 Fructose conversion to 5-HMF catalyzed by a) CTFs functionalized with the -SO₃H on the aromatic core; adapted from ref. 185, b) on the linking N atom; adapted from ref. 157, c) comparison between a HCP (sample S1) and a PAF (sample S2). Adapted from ref. 186.

homogeneous acids (*e.g.*, PTSA) and benchmark heterogeneous systems like Amberlyst-15.^{157,185,186} However, the specific advantages of each material class remain obscured by a lack of direct, standardized comparison. For instance, while high surface area is often cited, the comparative studies by Du *et al.* do not isolate this effect from other variables like acid strength or hydrophobicity.¹⁸⁶ Consequently, selecting the most suitable SPOP for a given application remains challenging. Future work must prioritize systematic, side-by-side evaluations under identical conditions and employ advanced techniques (*e.g.*, DFT analysis, *in situ* spectroscopy) to establish clear structure–activity relationships that link specific material properties such as pore hierarchy, acid site density and accessibility, and framework hydrophobicity to catalytic performance and stability.¹⁸⁷

Conversely, other biomolecules may be valorized too using SPOPs. For instance, Artz *et al.* employed sulfonated CTFs as catalysts in the hydrolysis of cellobiose to glucose.¹⁸⁸ The versatility and monomer engineering for CTF synthesis was exploited in this example. Indeed, they screened different

materials using monomers of varied complexity, *i.e.*, 1,3-dicyanobenzene, 1,4-diazobenzene, 2,6-dicyanopyridine and 4,4'-dicyanobiphenyl. The catalysts could achieve the selective synthesis of glucose in just 4 h of reaction, yields up to 86% at a very good substrate-to-catalyst ratio of 10. Their most active material used 1,3-dicyanobenzene, even though some sulfur leaching was observed during operation due to the catalytic hydrothermal conditions. On the other hand, confined Brønsted acidic ILs based on 1-methyl-3-(3-sulfopropyl)-1*H*-imidazol-3-ium hydrosulfate in the cavities of an iminic COF presented very good activity in the dehydrative formation of isosorbide from sorbitol.¹⁸⁹ Even though no covalent bonding between polymer and active site was created, the material resulted in the best catalyst under study, outperforming benchmark and homogeneous related catalyst, with a yield of isosorbide as high as 97% and good recyclability. H-bonding between the salt and the polymer, which possess the adequate scaffold to launch that interactions and the suitable morphology upon design of that COF were concluded to be the mechanistically consequence of



the robustness observed together with the contribution to the stabilization of the active site. In contrast, Yuan and coworkers synthesized a series of polydivinylbenzene-based solid acid catalysts (PDSF-*x*) with tunable acid strength by co-grafting $-\text{SO}_3\text{H}$ and strongly electron-withdrawing trifluoromethanesulfonic acid ($-\text{SO}_2\text{CF}_3$) groups onto a PDVB framework.¹⁹⁰ The variable “*x*” in PDSF-*x* denotes the amount of $\text{CF}_3\text{SO}_3\text{H}$ used in the functionalization. They investigated the effect of increasing acid strength on the solvent-free dehydration of sorbitol to isosorbide. Interestingly, while the total sulfur content (and thus acid site loading) increased from 1.75 to 3.73 mmol g^{-1} , the catalytic activity decreased, with initial turnover frequencies (TOFs) dropping from 231 h^{-1} to around 200 h^{-1} . The authors attributed this decline to pore blockage from excessive grafting, which reduced the specific surface area and made active sites less accessible. Consequently, the catalyst with moderate functionalization, PDSF-0.05, delivered the optimal performance, complete sorbitol conversion, an 80% yield of isosorbide at 140 °C, and excellent recyclability.

In summary, the performance of these catalysts is intrinsically governed by their material properties. A critical insight from comparing these works is that for SPOPs, the highest degree of functionalization (*e.g.*, sulfonation) does not automatically guarantee the highest catalytic activity or operational stability. Optimal performance arises from a balanced interplay of acidity, porosity, and structural robustness. To advance this field, systematic studies are needed to decouple these factors, correlating specific linker functionalities, pore architectures, and acid site densities with activity and durability. Furthermore, expanding the valorization of diverse organic residues into SPOPs represents a promising pathway for synthesizing a broader range of high-value green chemicals.

3.4. Biofuel synthesis: esterification and transesterification

Biofuel is a fuel produced from a short span of time from biomass.²²² The definition brings together different categories of chemicals, ranging from biologically produced methane and short-chain alcohols such as methanol, ethanol, propanol, and butanol (in their various isomers) to long-chain compounds commonly referred to as biodiesel. Indeed, biodiesel is commonly produced from oils and fats by transesterification reactions to convert into fatty acid methyl esters (FAMES).²²³ Since the formation of esters, including FAMES, is a carbonyl reaction typically catalyzed by acids, SPOPs are excellent candidates to carry out this reaction thanks to their intrinsic acidity. In addition, the advantages of specific and precisely located $-\text{SO}_3\text{H}$ groups within the polymer network, tunable design and heterogeneous characteristics towards separation after reaction make them appealing candidates for an important challenge as the production of fuels avoiding petroleum practices.

In this regard, SHCPs have been deeply studied, maybe due to their easy synthesis and facile functionalization. For instance, hollow nanosphere polymer particles were used by Song *et al.* to catalyze biodiesel formation. They synthesized spheric HCP particles ranging 500 nm in diameter that were sulfonated after synthesis with ClSO_3H , and they tested the methanolysis

(esterification with MeOH) of both lauric acid and coconut oil.¹⁹¹ Good results were achieved (yields higher than 85%) with all samples under study compared to the non-sulfonated material due to the acidity provided by the sulfonic groups as mechanistic reason. Guo and Liang prepared a hyper crosslinked xylene-based polymer doped with naphthalene to, according to authors design, provide more sulfonation sites.¹⁹² They achieved packed spheres configuration that were active in the tetrahydropyranylation of alcohols and in the transesterification of waste oil with methanol, achieving over 90% yield of free fatty acids in their reaction mixture after 4 h. The doped agent resulted beneficial for the functionalization-catalytic activity purposes. The materials presented excellent recyclability too thanks to the inherent robustness of the HCP structure. The tunability that presents HCPs was exploited by Schweng and coworkers (Fig. 8a).¹⁹³ Indeed, they synthesized four different materials varying the density of $-\text{SO}_3\text{H}$ groups and tested them in the model esterification of hexanoic acid with benzyl alcohol. They discovered that the higher the acidity the higher the catalytic activity, in contrast with the 5-HMF production discussed above. Indeed, a

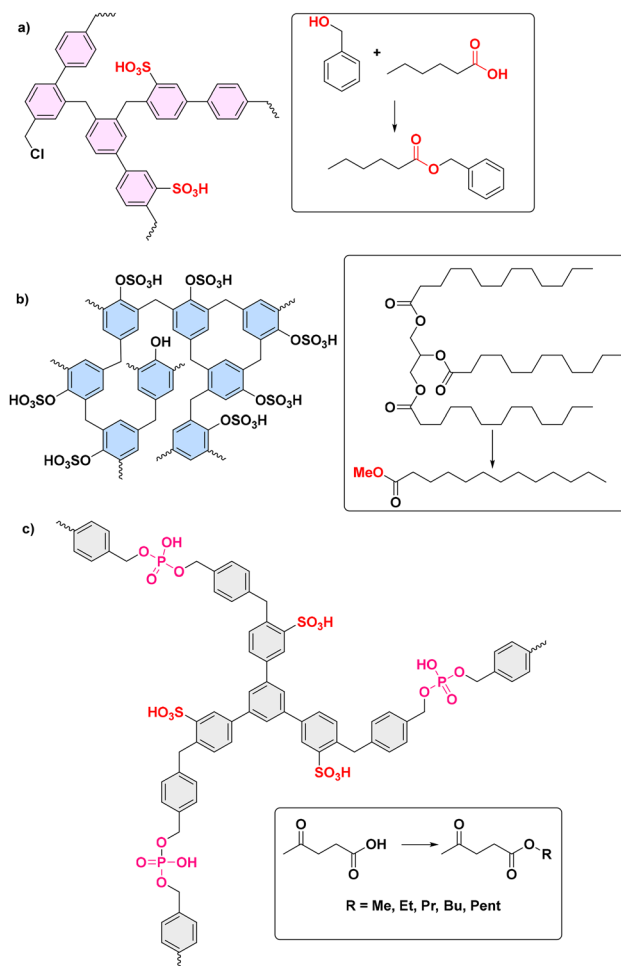


Fig. 8 Different esterification or transesterification reactions catalyzed by HCPs. a) HCP for the esterification of heptanoic acid with benzyl alcohol; b) polyphenol-based HCP for the synthesis of biodiesel; c) phosphate-linking polymer for the esterification of levulinic acid. Adapted from ref. 163, 193 and 194.



maximum was reached of $95 \pm 4\%$ after 22 h for the most acidic material. However, this high level of acidity led to uncontrolled selectivity compared to the other less acidic materials, where the target product was obtained, although with less yield, in high selectivity. Therefore, in stark concordance with the previous data, a balance must be met to develop efficient catalysts. In a similar approach, Bhunia *et al.* presented a supermicroporous HCP thanks to monomer design (FC polymerization of carbazole with α, α' -dibromoxylene) that was post-synthetically sulfonated.¹¹⁹ The material efficiently catalyzed the esterification with methanol of different fatty acids, such as oleic, palmitic and stearic among others, with enhanced performance compared to the non-sulfonated polymer. Biodiesel was also produced by a SHCP based on polyphenols (Fig. 8b).¹⁶³ The samples produced were tested in the esterification and transesterification of various long-chain fatty acids and fatty oils as soy or coconut oil. Authors discovered that catalytic activity boosted up with increasing acid functionalization, and a significant degree of sulfonation was required, being the most acidic the most active material, although selectivity and the effect of phenolic groups were not addressed. Furthermore, the materials presented good recyclability without leaching of acid sites. Summing up, SHCPs are outstanding candidates for catalyzing esterification reactions. Their suitability stems from straightforward synthesis, ease of functionalization *via* sulfonation, and the ability to precisely control the acid site density. They account accessible sulfonic groups that serve as heterogeneous acids prompted to catalyze the esterification mechanism, and the hydrophobicity of the material benefits substrate diffusion compared to homogeneous systems. Benchmarking against state-of-the-art catalysts confirms that SPOPs often deliver superior activity under significantly milder conditions, underscoring their transformative potential when judiciously designed. However, more insights into the actual mechanisms, including theory and experiment, are needed as in the precedent examples.

Other acid groups could also be employed to complement the sulfonic group acidity, such as the sulfonated porous polymer presented by Kim *et al.* that also contained a linking phosphate (Fig. 8c).¹⁹⁴ This material catalyzed the esterification of levulinic acid with aliphatic alcohols (from methanol to pentanol) with good recyclability and enhanced catalytic character compared to the non-sulfonated material, thus indicating the low acidity of the phosphate and the need of stronger acid sites. However, the role of the phosphate in the activity might require further research by advanced techniques to assess the viability of this strategy towards real applications. Finally, esterification reactions including acetic acid with cyclohexanol or fatty acids (hexanoic, lauric) with ethanol were efficiently catalyzed by a sulfonated mesoporous polydivinylbenzene, prepared *via* PSM using ClSO_3H .¹¹⁸ Therefore, other structures rather than HCP are compatible with the biofuel pool. Indeed, sulfonated COFs were also employed as catalysts in biodiesel formation. Jia *et al.* presented an iminic COF whose monomers contained the sulfonic group.¹⁹⁵ The acidity of the material was increased introducing in the pores the IL 1-methyl-3-(3-sulfopropyl)-1*H*-imidazol-3-ium hydrosulfate. This material presented a 91% conversion of

the model soybean oil into FAMES, which is far better than the 5% conversion achieved by the pristine material, and it also worked on real systems with good reusability. Surprisingly, the iminic COF resisted the acid mediated reaction, in a sign of inherent robustness that validates this kind of material for acid-mediated transformations. Many acid-catalyzed reactions involve hydrophobic organic reactants and suffer in aqueous or polar environments due to catalyst deactivation or poor substrate affinity. SPOPs offer a unique solution through inherent or engineered hydrophobicity of their organic backbones, which creates a favorable microenvironment for organic transformations. Munyentwali *et al.* highlighted this in their study of a rigid, hydrophobic POP for oleic acid esterification.¹¹⁶ They concluded that strong surface hydrophobicity was key to outperforming hydrophilic benchmarks like Nafion[®] NR50 and Amberlyst-15. The hydrophobic pores favored the partitioning of fatty acid reactants, leading to higher local concentrations at the $-\text{SO}_3\text{H}$ sites. This intrinsic property of many POPs provides a built-in advantage over hydrophilic solid acids in non-polar reaction media. On their work, Goesten *et al.* showed that a sulfonated PAF maintained comparable or superior initial activity to Amberlyst-15 in acetic acid esterification with *n*-butanol, but with minimal deactivation over multiple cycles.¹²² This contrasts with Amberlyst-15, which suffers from swelling and acid site leaching in polar media, highlighting the stability advantage of rigid SPOP frameworks.

In summary, the acid character of the sulfonated polymers allows the efficient esterification reaction to be performed to obtain biodiesel and related compound. However, characterization of the active sites in operando rather than invoking the acidic character as general mechanism may be key for definitively assessing the true nature of the active sites, synergies, structure-properties relationships and the actual benefits for employing SPOPs in this type of chemistry. The activity values are clear, but more is needed to unseat classic systems.

3.5. Polymerization

Innovative and more sustainable polymerization reactions or processes, that demonstrate enhanced activity, efficiency, control and selectivity continues to be key for the synthesis of important added value materials with specific chemistry and application.^{224,225} In this way, several synthetic processes in polymer technology are still catalyzed or mediated by PTSA.²²⁶ This strong acid activated the monomer (or the initiator) generating the active species that sustains polymer growth. Thus, sulfone-decorated polymers might have room for development in the search for more efficient technologies, where the heterogeneous character might have advantages such as more efficient reagent diffusion during polymerization to avoid reactor fouling and easy separation after reaction, combined with localized solid acid nature.

In this regard, Park *et al.* used a sulfonated microporous polymer as a catalyst for polyketone synthesis from CO and ethylene (Fig. 9a).¹²¹ The porous catalyst produced ~ 5 g of polymer per 1.5 mg of catalyst without fouling the reactor, a stark contrast to PTSA homogeneous system. The confined



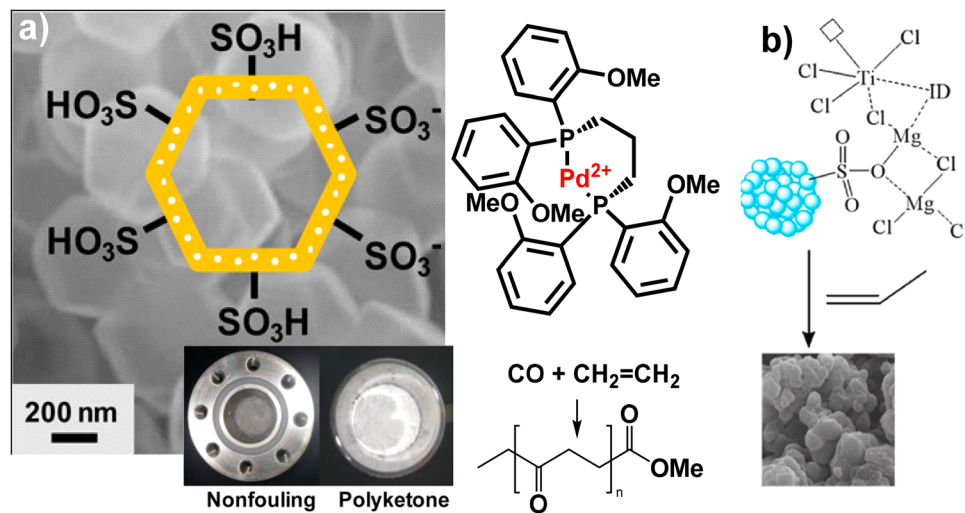


Fig. 9 Sulfonated porous polymers as a platform to sustain the polymerization of a) polyketones. Reproduced with permission from ref. 121 Copyright 2016 American Chemical Society and b) polypropylene with the Ziegler–Natta catalyst. Reproduced with permission from ref. 197 Copyright 2022 Elsevier.

pores likely control monomer access and polymer chain growth, demonstrating how SPOP architecture can solve practical engineering challenges in polymer production. Similar catalytic system was presented by Kang *et al.*¹⁹⁶ They coated another microporous organic polymer in silica particles and then, samples were sulfonated. The resulting material was employed as seed in the catalytic terpolymerization of CO, ethylene, and propylene. Under optimal conditions, they obtained 17.2 kg of polymer per gram without fouling, and the resulting polymers exhibited a narrower molecular weight distribution than those produced by homogeneous systems. In both cases, the SPOP synthetic method resulted in materials with the adequate porosity to yield the target polymers without pore blockage. Thus, templated sulfonated microporous organic polymers are excellent hybrid catalysts where monomers can easily find the active and diffuse after reaction because the sulfonation seemed to both host the metal complex and enhance the dispersion in the reaction medium, facilitating the polymerization. In a different approach without the hard template method, the stepwise protocol based on synthesis and sulfonation was also applied to yield a material by FC polymerization.¹⁴⁷ Therefore, self-synthesized SPOPs are also valid candidates to catalyze polymerizations too. This POP was tested in the dimerization of isopentane. Indeed, the alkylation and/or dimerization of alkanes is an important reaction since it provides products that can be directly introduced in the gasoline pool. This reaction is typically carried out both academic and industrially under acidic conditions, and that SPOP carried it out with good results: 24 h of reaction at 140 °C afforded the best results in terms of conversion, yield and selectivity (80%, 75% and 95%, respectively), with better activity than acidic zeolites compared in the study. The localized acidity in the SPOP may be responsible for the observed activity. Finally, the group of Wang presented two works dealing with the Ziegler–Natta process²²⁷ using CH_3MgCl and TiCl_4 as precursors to be added over the

sulfonated polymer.^{197,198} The reaction was classically performed over alumina substrates, but the SPOP may present some advantages. With this material, authors achieved the catalytic synthesis of ultrahigh molecular weight polyethylene, as well as polypropylene (Fig. 9b),¹⁹⁷ which exhibited both high stereocontrol and a broad polydispersity. Both works highlight the chemical environment of Ti and Mg sites due to the $-\text{SO}_3\text{H}$ groups present in the material (even coordinating the Ziegler–Natta centers), which resulted beneficial for the catalytic purposes. In contrast, conventional systems did not present this chemical ambient, thus highlighting the excellent performance of that SPOPs in industrially important reactions as Ziegler–Natta polymerizations.

However, few studies on SPOPs to sustain polymerizations have been carried out, in particular lacking more types of POPs. More investigations, both at synthesis and mechanism levels and experimental and theoretical approaches are required to introduce these soft materials into a realm dominated by doped inorganic systems, where milder synthesis and the intrinsic tunability of the SPOPs can compete for a more efficient polymer synthesis.

3.6. Cascade reactions

The acidic nature of porous frameworks can be employed to trigger cascade or tandem processes.^{228,229} Indeed, this section devoted to catalysis with sulfonated polymers has discussed how the solid acid character can catalyze different reactions, such as condensation, dehydrations, esterifications and polymerizations among others. Thus, a proper reaction design can exploit this character and include an acid-mediated step in a more complex overall reaction. On the other hand, the acid nature can be coupled with other active sites to trigger complex catalytic processes. Merging all the required parameters into a solid polymer, versatile enough to allow the synthesis upon design, and combined with the inherent robustness and easy separation after reaction would be quite beneficial for the



development of advanced heterogeneous catalysts. For cascade reactions, the confined environment of SPOPs pores can be essential for facilitating intermediate transfer. In their study, Gong and colleagues reported presented a sulfonated COF able to catalyze different reactions.¹⁹⁹ They obtained their catalyst by acid-mediated trimerization from a diketone containing a -OH group, and then the decoration was carried out by alkylation with 1,3-propanesultone (Fig. 10a). COFs demonstrate outstanding flexibility to allow this synthetic design and polymerization conditions. This material was able to catalyze the synthesis of benzimidazoles, benzodiazepines and one-pot reductive aminations, reactions that require an acid activation that was efficiently performed by the chosen SPOP. For the first process, the sulfonic material catalyzed the imine formation, which let the substrate ready for an intramolecular cyclization; thus, a broad scope of benzimidazoles with good activity (yields above 90% in 1 h reaction) was presented. The authors proposed that the periodic porous architecture of the COF does more than simply immobilize -SO₃H groups; it enforces proximity between reactants and catalytic sites, creating a local environment where reactive intermediates are stabilized and their intermolecular

diffusion is minimized. This overcomes the fundamental kinetic challenges of diffusion-controlled reactions and allows for the efficient, sequential execution of multiple catalytic steps, a cooperative synergy difficult to replicate with a physical mixture of homogeneous catalysts, which often suffer from incompatibility and low local concentrations of intermediates. A similar strategy was designed for the benzodiazepine synthesis, and another broad scope was studied with very good catalytic activity. Finally, in the presence of reductant that did not affect the structure of the catalyst thanks to the intrinsic COF robustness, an imine and an aldehyde were condensed and reduced in a one-pot process. However, the proposed mechanism just stays in a simple acid activation of the substrates. In a different work, Kalita *et al.* synthesized a -SO₃H functionalized tetraphenylethylene-based HCP featuring a well-developed porous network and accessible -SO₃H sites.²⁰⁰ The yielded SHCP was tested in the ambitious synthesis of symmetrical triaryl-methanes *via* a dual C-C bond cleavage. Good activity in terms of high yields (33–94%) was reported. This reaction is typically performed by strong and aggressive Lewis acids as FeCl₃, but authors could outperform this system with their polymer. The

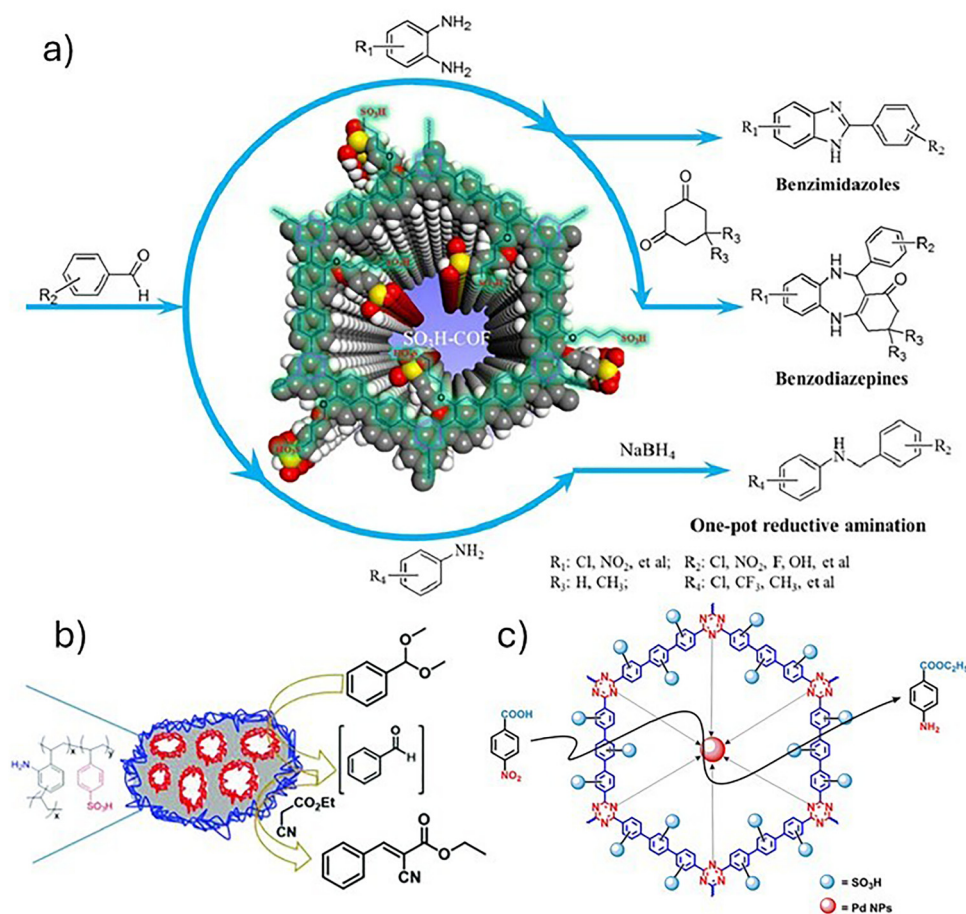


Fig. 10 Cascade processes catalyzed by -SO₃H decorated POPs. a) Synthesis of benzimidazoles, benzodiazepines and reductive amination by a sulfonated COF. Reproduced with permission from ref. 160 Copyright 2022 Elsevier; b) acetalization and Knoevenagel condensation using modified HCP. Reproduced with permission from ref. 197 Copyright 2019 Royal Society of Chemistry; c) reduction and esterification with a Pd-sulfonated CTF. Reproduced with permission from ref. 162, 199 and 201 Copyright 2021 Elsevier.



scope of method was also explored with different substrates varying their characteristics. In addition, recovery of the catalyst by filtration allowed five consecutive reuse cycles without observing decrement in the activity, overcoming industrial limitations in terms of separation and recovery that the homogeneous catalyst cannot present. The material prepared by Sun and colleagues enabled two successive acid–base mediated reactions.¹⁶² Indeed, they performed the polymerization of DVB in the presence of a monotopic sulfonate, which was further modified with an amine and acidified to provide a bifunctional catalyst. Thus, the resulting POP was acidic enough to break an acetal and basic enough to condensate the resulting aldehyde with a malonate derivative (Fig. 10b). Only this bifunctional material accomplished the cascade process, in contrast to pristine, sulfonated polymer or homogeneous catalysts that lacked the required active sites. Therefore, the aromatic core of the selected SPOP was reactive enough to support the intended functionalizations that led with the observed catalytic activity. Corma *et al.* demonstrated that the bifunctional PAF catalyst PPAF-SO₃H-NH₂ achieved 100% yield in a one-pot cascade reaction involving acetal hydrolysis followed by Knoevenagel condensation, far surpassing mixtures of its monofunctional counterparts or commercial resins. This result confirms that the proximity and cooperative action of acid and base sites within a single, mesoporous framework are essential for high-efficiency cascade catalysis.²²⁹ A transformative frontier for SPOPs is the integration of -SO₃H sites with other functional groups (*e.g.*, metals, basic sites) within a single framework to enable cooperative catalysis. This creates bifunctional or multifunctional catalysts capable of driving sophisticated tandem processes. A prime example is the work by Raza *et al.*, who fabricated a Pd nanoparticle-loaded sulfonated CTF.²⁰¹ This material catalyzed a one-pot

reduction-esterification cascade, converting 4-nitrobenzoic acid to benzocaine (Fig. 10c). Control experiments proved both functionalities were essential: Pd nanoparticles reduced the nitro group, while the proximal -SO₃H sites esterified the resulting acid. The capacity of SPOP frameworks to spatially organize disparate active sites such as proximal -SO₃H groups and metal nanoparticles, enables efficient telescoped syntheses in a one-pot operation, circumventing the need for intermediate separation and re-catalysis. Current literature predominantly shows two-step cascades, where the designed acidity and hydrolytic stability of the polymer are critical. While this demonstrates the successful integration of acid activation with other functions (*e.g.*, redox metals, basic sites), the field must now advance toward longer, more complex catalytic sequences. Crucially, a deeper mechanistic understanding, elucidating how substrates interact with the confined pore environment and co-localized sites, is essential to rationally design SPOPs that fully unleash their potential for precise multi-step catalysis.

Asymmetric catalysis is an emerging field in the realm of organic materials,^{230–232} and it was also attempted using -SO₃H functionalized organic polymers. Indeed, Pang *et al.* constructed a β -ketoenamine COF with -SO₃H groups that were further functionalized with a chiral diamine to create a chiral sulfonamide catalyst.²⁰² This material catalyzed an asymmetric Michael addition of acetone to β -nitrostyrene with high efficiency of ~80% and enantiomeric excess (ee) of 90%. For a series of β -nitroolefins, enantioselectivities ranged from 84% to 96%. The proposed mechanism involves a synergy where the chiral center guides enantioselection while the free -SO₃H group activates the substrate *via* acid–base interaction (Fig. 11). Although current systems may face stability limitations, this work pioneers the use

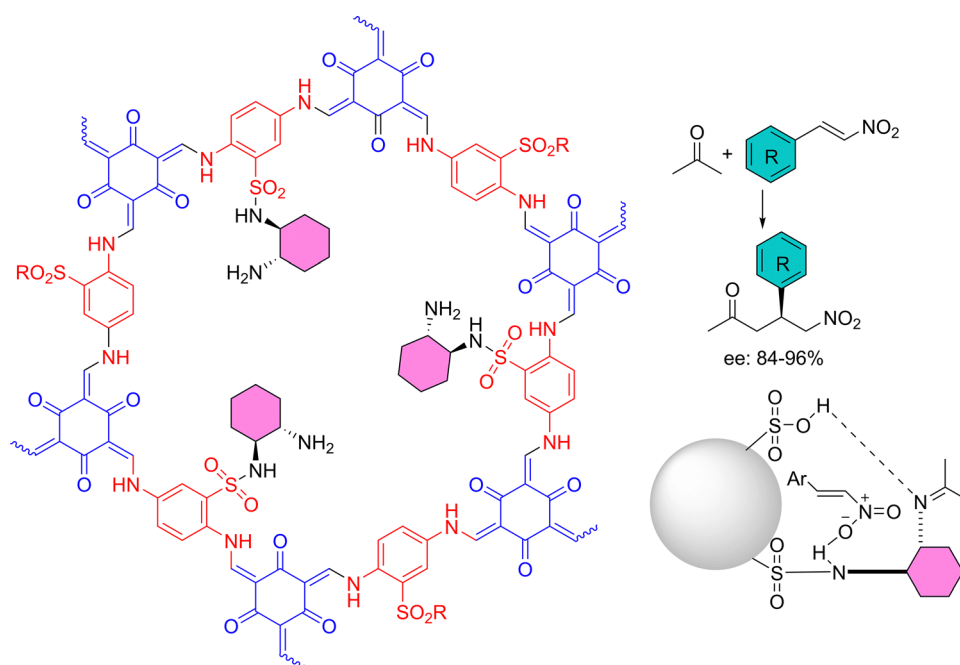


Fig. 11 Enantioselective Michael addition with a proposed mechanism using sulfonated-functionalized β -ketoenamine porous polymer (R = H or organocatalyst). Adapted from ref. 202.



of SPOPs as modular platforms for heterogeneous enantioselective acid catalysis.

4. Environmental remediation

The incorporation of $-\text{SO}_3\text{H}$ groups into POPs represents a powerful strategy to engineer high-performance adsorbents for environmental remediation. This approach leverages a synergistic interplay between the intrinsic properties of the porous scaffold and the introduced functional motifs. The inherent high surface area and permanent porosity of POPs facilitate mass transfer and physical entrapment of pollutants *via* pore-filling. Concurrently, sulfonation serves a dual purpose: (i) it drastically improves the hydrophilicity and aqueous dispersibility of otherwise often hydrophobic frameworks, ensuring optimal contact with waterborne contaminants, and (ii) it decorates the pore walls with a high density of anionic sulfonate ($-\text{SO}_3^-$) groups, enabling strong, specific electrostatic interactions with cationic organic pollutants (*e.g.*, dyes, antibiotics) and facilitates ion-exchange processes, wherein protons from the $-\text{SO}_3\text{H}$ groups are exchanged for cationic metal ions. Furthermore, the aromatic backbones prevalent in many POPs contribute through π - π stacking interactions with pollutant molecules, while hydrogen bonding can also be a contributing factor. The culmination of these complementary mechanisms including electrostatic attraction, ion exchange, π - π interactions, hydrogen bonding, and pore filling often results in exceptional adsorption performances, characterized by remarkably fast kinetics and capacities that can exceed 1000 mg g^{-1} . However, a critical challenge in designing such materials lies in balancing the degree of sulfonation with the preservation of porosity, as the grafting of bulky $-\text{SO}_3\text{H}$ groups can lead to pore blocking and a significant reduction in surface area, a common trade-off that impacts overall adsorption performance. Table 4 summarizes the use of SPOPs for environmental remediation including dye adsorption, antibiotic/drugs removal, metal ion adsorption, and gas separation.

4.1. Dye adsorption

Incorporating $-\text{SO}_3\text{H}$ groups into POPs is a highly effective method for enhancing the adsorption of dyes (Scheme 9), with a pronounced effect on cationic dyes.

This approach activates multiple complementary mechanisms. The innate high surface area and porous structure of POPs provide a foundation for physical entrapment through pore filling. Sulfonation, on the other hand, introduces a strong electrostatic force that captures cationic dyes. Additionally, the resulting anionic frameworks demonstrate increased hydrophilicity, which improves their dispersion in water and allows better access for dyes to the internal pore surfaces. The combination of this electrostatic attraction, the inherent π - π stacking ability of the aromatic polymer backbone, and the high porosity creates a synergy that leads to remarkably fast adsorption kinetics and capacities. In many cases, these performance metrics surpass those of unfunctionalized POPs and numerous benchmark adsorbents. A notable example is the synthesis of a sulfonate-grafted tetraphenylethylene-based HCP

(SHCTPE) *via* molecular knitting and post-sulfonation (Fig. 12).²³³ While the sulfonation step reduced the BET surface area from $1561 \text{ m}^2 \text{ g}^{-1}$ (precursor HCTPE) to $860 \text{ m}^2 \text{ g}^{-1}$, it conferred superhydrophilicity and a negatively charged skeleton, leading to exceptional, ultrafast adsorption kinetics for MB and RhB with $>99\%$ removal in just 5 seconds and pseudo-second-order rate constants of 58.7 and $83.4 \text{ g mg}^{-1} \text{ min}^{-1}$ for MB and RhB, respectively. Furthermore, the adsorption capacities achieved for MB and RhB were 1410 mg g^{-1} and 1096 mg g^{-1} , respectively. SHCTPE also retained its performance over five cycles, demonstrating excellent stability. This serves as a promising example of a superhydrophilic HCP for the efficient adsorption of organic dyes in aqueous solutions. This underscores that for rapid aqueous-phase adsorption, enhanced hydrophilicity and electrostatic driving force can outweigh the importance of ultra-high surface area.

In a subsequent study, a sulfonated COP (TC-COP- SO_3H) (Fig. 13) derived from triphenylmethane and cyanuric chloride experienced a more drastic loss in porosity (Langmuir surface area dropped from 137 to $50 \text{ m}^2 \text{ g}^{-1}$ upon sulfonation), yet still achieved enhanced capacities for BB-41 and -46 compared to its non-sulfonated counterpart.²³⁴ This improved performance of TC-COP- SO_3H is due to the high density of adsorptive sites, which facilitate favorable electrostatic interactions with cationic dye molecules, compensating for its significantly reduced porosity.

A significant methodological advance to circumvent the porosity-loss dilemma was demonstrated by Yang *et al.*,²³⁵ who emphasized the critical role of the polymer's swelling state during modification. By employing a one-pot strategy where the PAF precursor remained in a fully swollen state before sulfonation, they synthesized PAF-215. This material achieved an impressive balance of a high sulfur content ($13.2 \text{ wt}\%$), indicating dense functionalization, and a retained BET surface area of $580 \text{ m}^2 \text{ g}^{-1}$. Consequently, PAF-215 exhibited top-tier adsorption capacities for RhB (1075 mg g^{-1}), MB (1020 mg g^{-1}), and antibiotics including tetracycline (TC, 826 mg g^{-1}) and ciprofloxacin (CIP, 1134 mg g^{-1}), showcasing that preserving pore accessibility during sulfonation is key to maximizing overall adsorbent performance. This exceptional adsorption performance is attributed to its hydrophilicity, high density of $-\text{SO}_3\text{H}$ groups, substantial specific surface area, and hierarchical pore structure.

Similarly, for inherently hydrophobic CMPs, sulfonation is essential to impart water dispersibility. To overcome these challenges, sulfonate-grafted CMPs were prepared with high dispersity in water (Fig. 14). The sulfonate-grafted pyrene-based CMPs, for instance, achieved a high MB capacity of 1650 mg g^{-1} , primarily attributed to the combined effects of electrostatic attraction and π - π stacking within their ordered porous structure.

In a recent contribution, an anionic sulfonate-grafted COF (TpStb- SO_3Na) was synthesized from Tp (Tp: 2,4,6-triformylphloroglucinol) and Stb- SO_3H (Stb- SO_3H : 4,4'-diaminostilbene-2,2'-disulfonic acid) through the Schiff-base condensation.¹⁵⁶ The framework features densely aligned sulfonate groups within its periodic one-dimensional channels (Fig. 15), which impart a



Table 4 Environmental remediation using SPOP: Adsorption, separation and removal

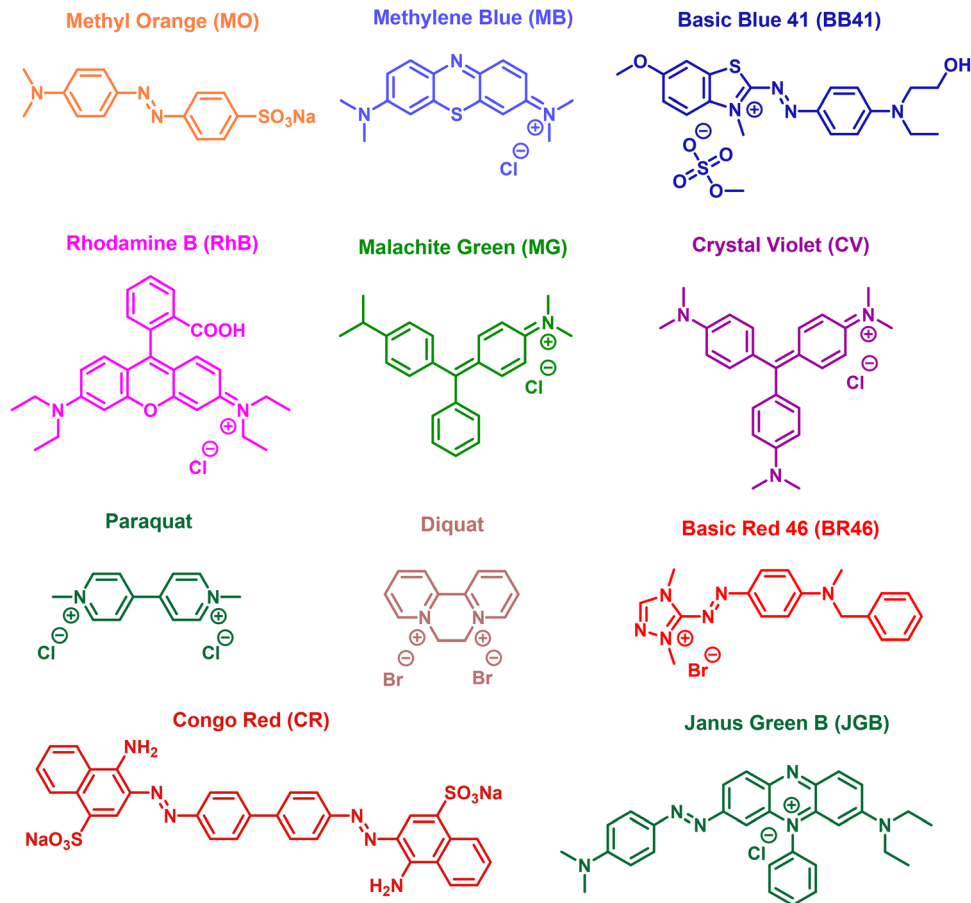
POP	BBs/monomers	Functionalization	BET surface area (m ² g ⁻¹) Pore size (nm)	Application	Adsorption capacity and performance	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
SHCTPE	HCTPE	FC polymerization	860/1.734	Adsorption of organic dyes in aqueous solution	99% removal after 5 s. 1410 mg g ⁻¹ for MB and 1096 mg g ⁻¹ for RhB	NR	Five cycles	No	233
TC-COP-SO ₃ H	Triphenylmethane and cyanuric chloride	FC reaction	Langmuir surface area of 50.15	Adsorption of cationic dyes	897.5 mg g ⁻¹ for basic blue 41 and 862.5 mg g ⁻¹ for basic red 46	NR	Three cycles and TGA	No	234
PAF-215	DVB	Conventional PSM	580/1.19	Adsorption of cationic dyes and antibiotics	1075 mg g ⁻¹ for RhB, 1020 mg g ⁻¹ for MB, 826 mg g ⁻¹ for TC and 1134 mg g ⁻¹ for CIP	13.2 wt%	Five cycles	No	235
Sulfone-grafted CMPS (SPTP and CPTP)	1,3,6,8-Tetrabromopyrene	Sulfone grafted PSM	SPTP - 290.3/- CPTP - 391.3/-	Adsorption of cationic dyes	1650 mg g ⁻¹ and 97% removal efficiency of MB (100 mg L ⁻¹) in 25 min	NR	N ₂ adsorption and desorption studies, TGA	No	130
TpStb-SO ₃ Na	Tp and Stb-SO ₃ H	Condensation reaction	15.4/2.65	Adsorption of organic pollutants from wastewater	For CV - 1861 mg g ⁻¹ , MB - 1078 mg g ⁻¹ , MG - 5857 mg g ⁻¹ , and JGB - 1339 mg g ⁻¹	NR	PXRD, FT-IR, N ₂ adsorption and desorption studies, TGA	No	156
HCP	Benzene and FDA	FC reaction	855/0.51	Dye separation	m-HCP-5 - CR from aqueous media 465 mg g ⁻¹	NR	N ₂ Sorption isotherm	No	236
H-S-CMPs	1,3,5-Triethylbenzene and 2,7-dibromocarbazole	Songashira-Hagihara cross-coupling	105.79/-	Adsorption of antibiotics and cationic dyes from wastewater	RhB - 206.2 mg g ⁻¹ , MB - 324.7 mg g ⁻¹ , CIP - 222.2 mg g ⁻¹ , norfloxacin (NOR) - 216.9 mg g ⁻¹	—	Five cycles, FT-IR, XPS, SEM	—	238
BaPy-SO ₃ H	Resorcinarene-based BaPy	Suzuki-Miyaura coupling reaction	3.07/0.10	Adsorption of organic pollutants	1850 mg g ⁻¹ for RhB, 2295 mg g ⁻¹ for MB, 1503 mg g ⁻¹ for CV, and 1304 mg g ⁻¹	—	SEM, Elemental mapping, N ₂ Isotherm curves, PXRD, IR	—	56
TH-SMONs	Tetra(4-ethylphenyl)methane with 1,4-diodobenzene	Sonagashira coupling	43/0.26	Drug delivery	—	6.14 wt%	Ten cycles, TGA, FT-IR	—	239
SHCPs	Triphenylmethane	FC reaction and PSM	939/3.7-3.9	Adsorption of antibiotic pollutants from water	427.5 mg g ⁻¹ of CIP	3.82 wt%	IR	—	240
TP-PPOs-SO ₃ H	TPP & triptycene	Conventional method	368.44/1.48	Adsorption of MB and CIP	MB - 1283.33 mg g ⁻¹ and CIP - 485.39 mg g ⁻¹	—	Eight cycles, SEM, FT-IR	—	241
Am-COF-SO ₃ H	Im-COF-SO ₃ H	1-ethyl-3-methylimidazolium tetrafluoroborate (Emim BF ₄) mediated oxidation	8.79/1.48	Adsorption of fluoroquinolone antibiotics in water	Enrofloxacin (ENR) - 614 mg g ⁻¹ , enoxacin (ENO) - 615 mg g ⁻¹ , NOR - 579 mg g ⁻¹	—	Six cycles, FT-IR, XPS, PXRD	—	241
SHCP	4,4'-bis(chloromethyl)-1,1'-biphenyl (BCMB)	One pot chemical synthesis	704/-	Removal of antibiotic pollutants	CIP - 757.7 mg g ⁻¹	12.4 wt%	Seven cycles, ATR-IR	—	166
TFPB-DABDA iCOFs	1,3,5-tris(<i>p</i> -formylphenyl)-benzene (TFPB) and 2,5-diaminobenzene-1,4-disulfonic acid (DABDA)	Schiff base reaction	39,80/3.769	Adsorption and separation of lanthanide compounds	The maximum adsorption capacity of early lanthanides can reach up to 0.1554 mmol g ⁻¹	—	—	—	242
SHCP-P-X X = 0.5, 1.5, 3.0, 4.0	4,4'-Bis(chloromethyl)-biphenyl (BCMBP)	Chemical synthesis	1044.6/-	Recover rare earth elements from tailing wastewater	106.78 mg g ⁻¹ (La), 111.99 mg g ⁻¹ (Eu) and 126.27 mg g ⁻¹ (Lu)	3.27 mmol g ⁻¹	Ten cycles, IR, BET	Yes	243





Table 4 (continued)

POP	BBs/monomers	Functionalization	BET surface area (m ² g ⁻¹) Pore size (nm)	Application	Adsorption capacity and performance	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
TAS-COF	2,4,6-triformylphloroglucinol (Tp) and 3,7-diaminodibenzol[<i>b</i> , <i>d</i>]thiophene (DAS)	Schiff base condensation	—	Colorimetric detection of UO ₂ ²⁺	LOD = 0.07 μmol L ⁻¹	—	SEM, EDS mapping, FT-IR, PXRD	—	244
TFPOTDB-SO ₃ H	DABSA and 2,4,6-tris-(4-formylphenoxy)-1,3,5-triazine	Solothermal reaction	190.73/1.55	Adsorption of As(III) from water	344.8 mg g ⁻¹	—	Four cycles, SEM, TEM, PXRD, N ₂ Sorption isotherm	—	245
SHCP	Fluorene-9-bisphenol (BHPF)	FC reaction	427.6/0.235	Adsorption of rubidium and cesium	For Ru ⁺ - 121.3 and Cs ⁺ - 133.9 mg g ⁻¹	—	Six cycles, SEM, FT-IR, PXRD, N ₂ Sorption isotherm, TGA	—	246
Composite aerogel	Sulfonated CMP (sCMP) and carboxymethyl cellulose (CMC)	physical doping and freeze-drying	16.737/0.028425	Adsorption of Pb ²⁺ and MB.	Pb ²⁺ - 94.93 and MB - 294.84 mg g ⁻¹	—	FT-IR, XPS, TGA, PXRD, SEM, EDS mapping	—	247
PIMS	Hexaphenylbenzene (HPB), triphenylbenzene (TPB), spirobisfluorene (SBF), triptycene (TRIP)	FC Polymerization	1585/-	CO ₂ adsorption and gas separation of N ₂ and CH ₄ from CO ₂	298 mg g ⁻¹	—	FT-IR, SEM, N ₂ adsorption isotherms	—	136
TpPa-SO ₃ Cu _{0.5}	Triformylphloroglucinol (Tp), DABSA (Pa-SO ₃ H)	Mechanical grinding synthetic method	85.98/1.3	Adsorption of NH ₃ nitrogen from water	30.45 mg N g ⁻¹	—	Five cycles, PXRD, XPS, FT-IR, TGA	—	248
1TCS@PDMS10	Sulphonated TCS	Post-oxidation and post-sulfonation Polymerization	72.5/-	NH ₃ capture	Up to 1.41 mmol g ⁻¹	—	ATR-IR, PXRD, SEM, Elemental analysis	—	249
Sulfonated polyimide (SPI)	ODA, BTDA	Polymerization	—	Methane storage as methane hydrate	Storage capacity 119 V/V,	—	Six cycles, FT-IR, SEM	—	250
PPN-6-SO ₃ Li	PPN	Sulfonation and lithiation	1186/-	CO ₂ adsorption	adsorption selectivity for CO ₂ over N ₂ (155 and 414 for PPN-6-SO ₃ H and PPN-6-SO ₃ Li, respectively)	—	N ₂ Sorption isotherm, SEM, Elemental analysis	—	53
SHCP-1	DCX/BDM/BCMBP	FC alkylation and PSM	940.33/3.96	Capture of NH ₃ from exhaust gases	11.54 mmol g ⁻¹	3.38 wt%	Ten cycles, N ₂ Sorption isotherm, SEM, Elemental analysis, IR, XPS	Yes	251
SPIM-1	5,5',6,6'-Tetrahydroxy-3,3',3',3'-tetramethyl-1,1',1'-spirobisindane (TTSBI) and 1,4-dicyanotetrafluorobenzene (DCTB)	Sulfonation using SO ₃ solution	642/-	Gas separation	Selectivity of H ₂ /N ₂ and O ₂ /N ₂ were 125 and 8.43 coupled with H ₂ and O ₂ permeability of 1077 and 73.4 Barrer resp	4.77 wt%	FT-IR, XPS, SEM/EDS mapping,	—	135



Scheme 9 Representative synthetic dye molecules to assess the adsorption capacities of SPOPs, including methyl orange (MO), methylene blue (MB), basic blue 41 (BB41), rhodamine B (RhB), malachite green (MG), crystal violet (CV), paraquat, diquat, basic red 46 (BR46), congo red (CR), and janus green B (JGB).

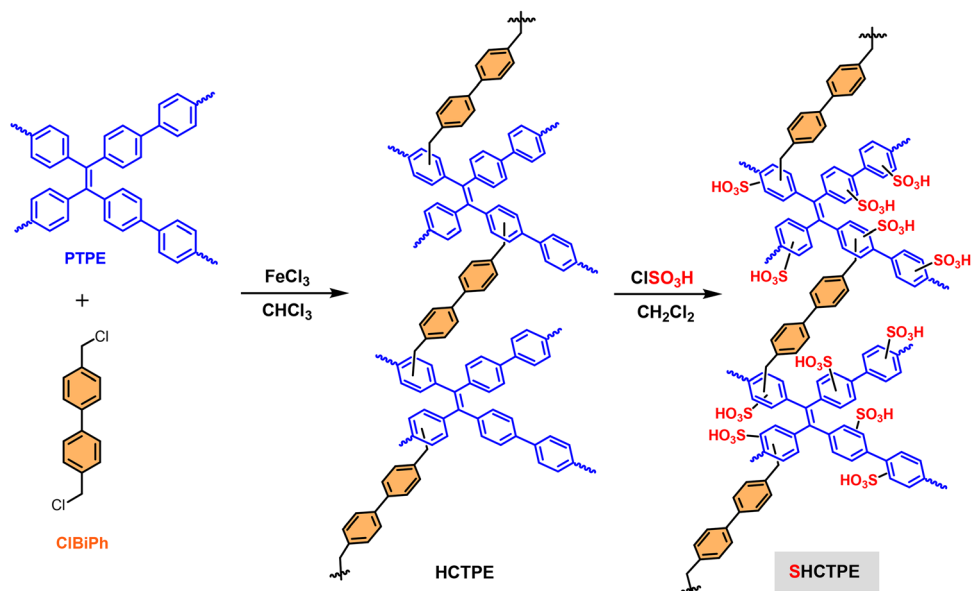


Fig. 12 Synthesis of HCTPE and SHCTPE solids. Adapted from ref. 233.



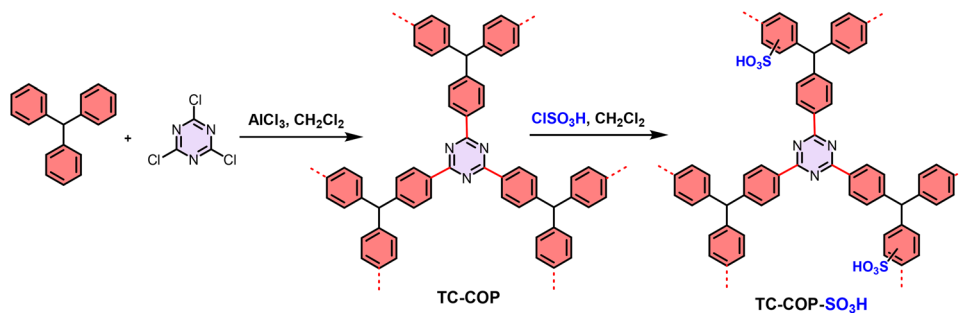


Fig. 13 Preparation process of TC-COP-SO₃H. Adapted from ref. 234.

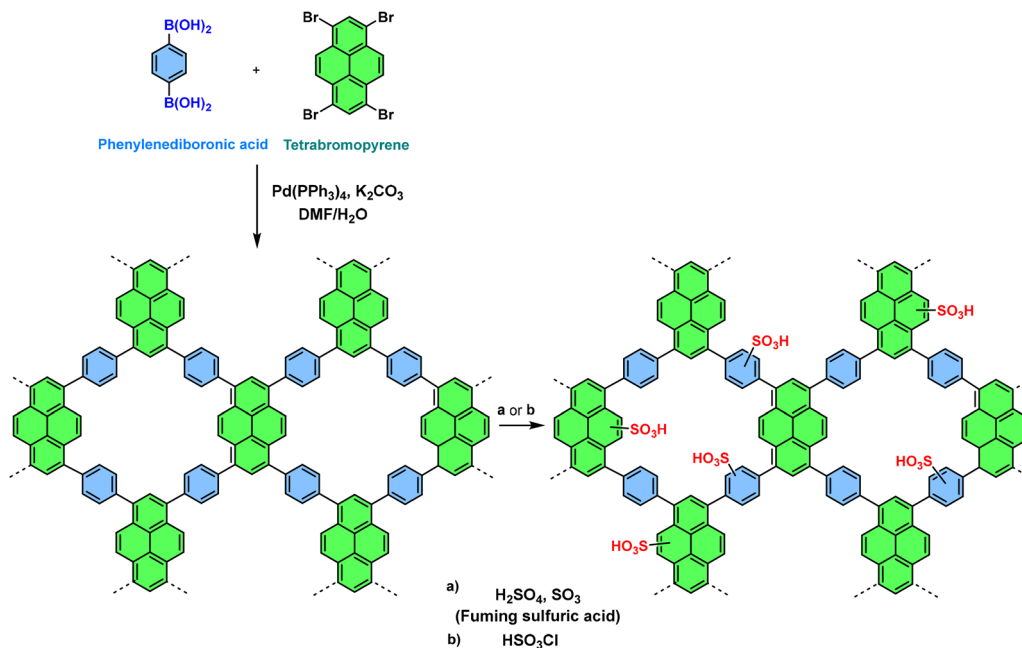


Fig. 14 The synthetic route of sulfonated-CMP. Adapted from ref. 130.

strong negative surface charge and enable ultrahigh adsorption capacities for cationic dyes.

Remarkably, TpStb-SO₃Na achieved record-high adsorption values for CV (1861 mg g⁻¹), MG (5857 mg g⁻¹), and JGB (1339 mg g⁻¹), surpassing all previously reported COFs, MOFs, and benchmark adsorbents.^{252–255} It also exhibited rapid kinetics (~99% uptake within 10 min) and high capacity for MB (1078 mg g⁻¹) at room temperature. The exceptional performance is attributed to the synergy of electrostatic attraction between the anionic sulfonate sites and cationic dye molecules, coupled with π - π interactions between the conjugated COF skeleton and the aromatic dye structures. Furthermore, TpStb-SO₃Na demonstrated excellent selectivity for cationic dyes over anionic ones and could be regenerated and reused for five cycles without significant loss of adsorption capacity, underscoring its potential for practical wastewater remediation.

In a complementary study, HCP membranes (m-HCPs) were fabricated *via* a versatile interfacial polymerization (IP) approach using benzene as the monomer (Fig. 16).²³⁶ The resulting m-HCP-5 membrane exhibited a high BET surface area of ~760 m² g⁻¹ with

a micropore-dominant structure. In dye-removal tests, m-HCP-5 showed an adsorption capacity of ~460 mg g⁻¹ for CR, while filtration experiments revealed a high-water flux of 55 L m⁻² h⁻¹ bar⁻¹ and anionic dye rejections exceeding 93%. The separation performance stems from the combined effects of size sieving (micropores ~0.5–1.2 nm) and electrostatic repulsion provided by the negatively charged sulfonate groups introduced during the interfacial polymerization process. This work highlights the potential of engineered porous organic membranes for simultaneous adsorption and charge-selective separation in advanced water-treatment applications.

In another precedent, a CMP possessing a hollow sphere was fabricated by using a template, SiO₂ nanoparticles and 1,3,5-triethynylbenzene and 2,7-dibromocarbazole as building blocks.²³⁷ Subsequently, hydrophilic CMPs were obtained by a sulfonation modification (H-S-CMPs) with better dispersibility in water. The performance of H-S-CMPs was investigated in the adsorption of dyes and antibiotics. The maximum adsorption capacities using H-S-CMPs were 206.2, 324.7, 222.2, and 216.9 mg g⁻¹ for RhB, MB, CIP, and norfloxacin, respectively.



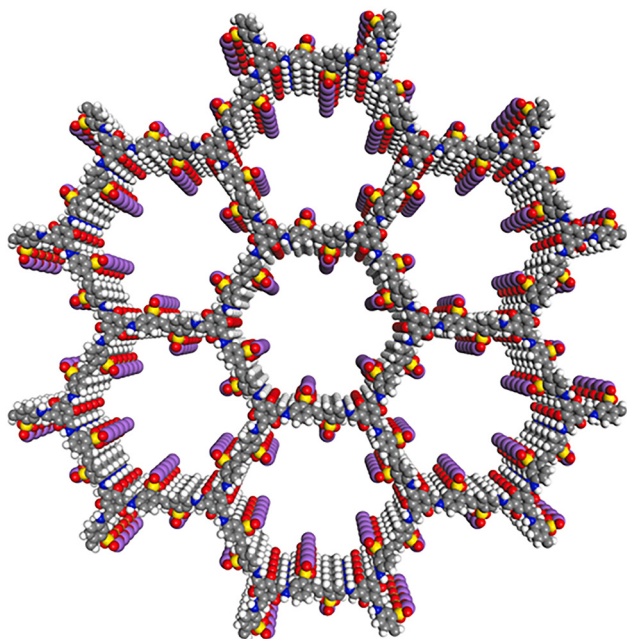


Fig. 15 Schematic of the TpStb-SO₃Na structure in an AA stacking motif. Reproduced with permission from ref. 156 Copyright 2023 Elsevier.

In addition, the adsorption mechanism of H-S-CMPs was due to the synergism between hydrogen bonding, electrostatic forces, π - π stacking and pore filling. The adsorption capacity of

H-S-CMPs remained stable after 5 cycles, and the removal rate of dyes exceeded 70%. A resorcinarene-based POP (BaPy) was designed and subsequently sulfonated to produce BaPy-SO₃H (Fig. 17).²³⁸ BaPy-SO₃H demonstrates exceptionally rapid adsorption rate, achieving equilibrium within just 1 min for a variety of organic pollutants (dyes, herbicides, and antibiotics) due to the establishment of electrostatic interactions, π - π interactions, and host-guest effects. BaPy-SO₃H showed the apparent rate constants of 31.24, 31.27, and 30.63 g mg⁻¹ min⁻¹ for CV, MG, and paraquat, respectively, which are the highest performance among reported adsorbents to the best of our knowledge.²⁵⁶⁻²⁶¹ Additionally, the BaPy-SO₃H exhibited impressive maximum adsorption capacities of 1850, 2295, 1503 and 1304 mg g⁻¹ for RhB, MB, CV and trimethoprim, respectively.

4.2. Antibiotic/pharmaceutical adsorption

Sulfonate groups are essential for the adsorption of antibiotic molecules, many of which have cationic functional groups under ambient conditions. The electrostatic attraction between the anionic sulfonate sites and the cationic parts of antibiotics like fluoroquinolones is a key mechanism for their effective removal from water, often resulting in high adsorption capacities and selectivity even in complex water conditions.

In one earlier report, a sulfonated MON (SMON) with triple- (TH-SMON) and double-shelled (DH-SMON) hollow spheres were synthesized, and their drug delivery performance was evaluated against that of single-shelled hollow sulfonated MON spheres

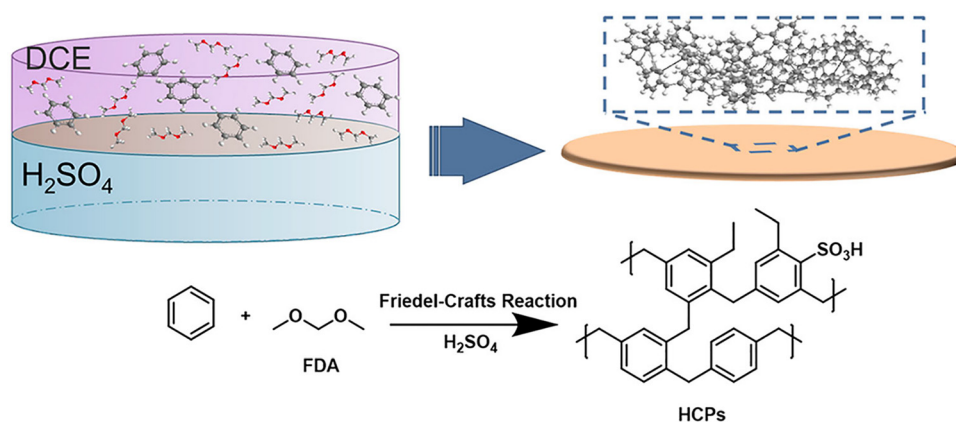


Fig. 16 Schematic diagram of m-HCPs fabrication with benzene as the monomer and formaldehyde dimethyl acetal (FDA). Reproduced with permission from ref. 236 Copyright 2024 Nature Publishing Group.

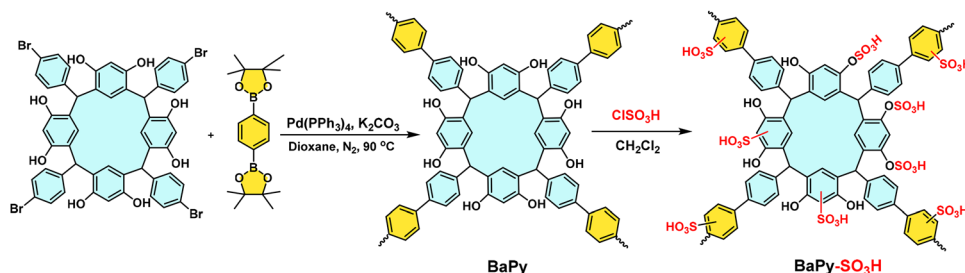


Fig. 17 Synthetic procedure of BaPy-SO₃H. Adapted from ref. 238.



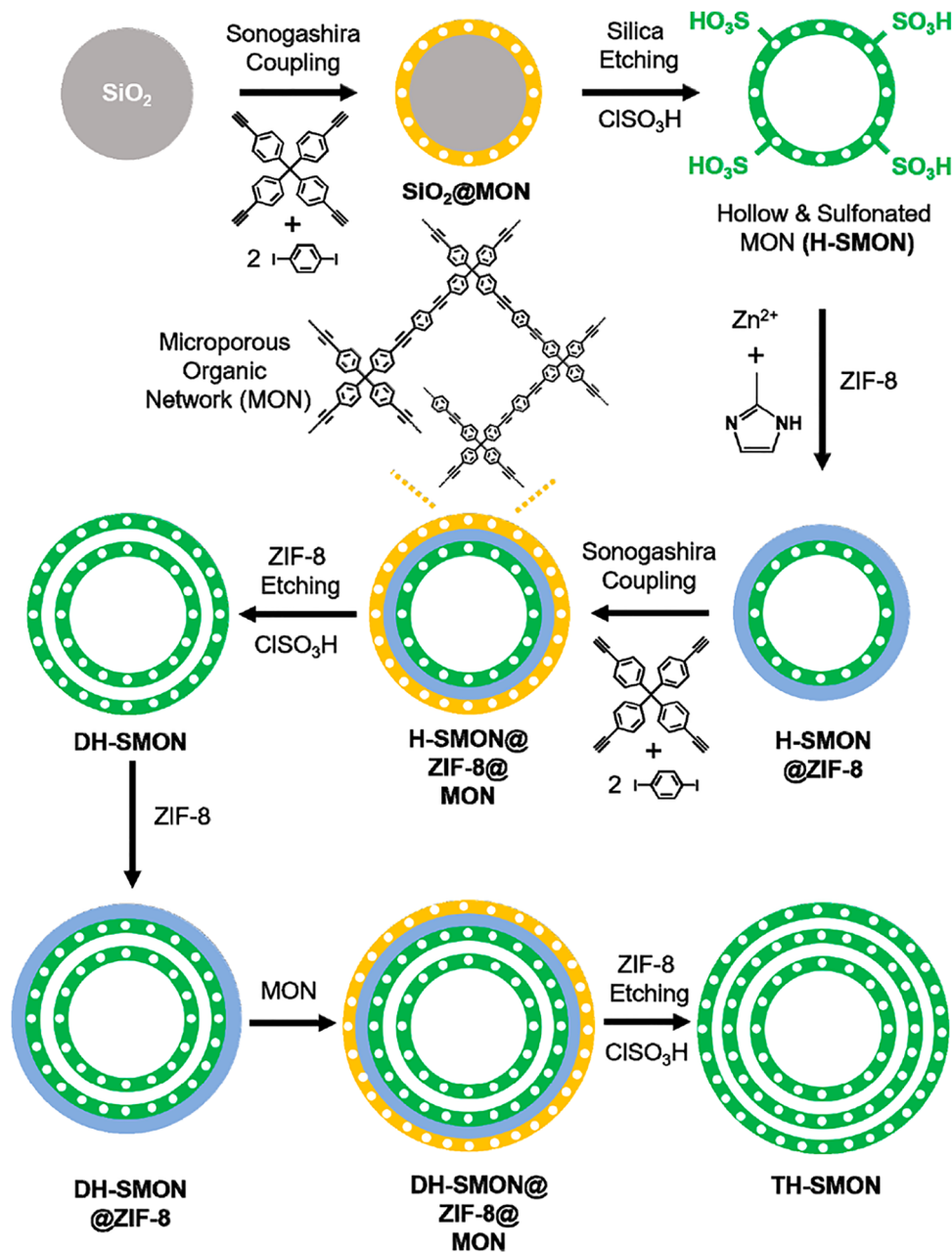


Fig. 18 Schematic outlining the synthesis of H-/DH-/TH-SMONs. Reproduced with permission from ref. 56 Copyright 2019 American Chemical Society.

(H-SMON) (Fig. 18).⁵⁶ The study revealed that both drug loading capacity and release efficiency generally increased with the number of shells. Using RhB as a model molecule, the loading capacity was achieved for H-SMON, DH-SMON and TH-SMON with 0.7 ± 0.3 wt%, 6.8 ± 0.2 wt% and 7.3 ± 0.2 wt%, which correspond to the loading efficiency of $61 \pm 3\%$, $73 \pm 3\%$, and $78 \pm 2\%$, respectively. Similarly, the cumulative release of RhB in PBS at pH 7.4 after 7 days at 37 °C increased as a function of thickness, from 41% to 60% from H-SMON to TH-SMON, respectively. The observed performance trend can be attributed to an expansion of the intershell space, enabling more facile infiltration of the aqueous electrolyte. All materials demonstrated excellent and stable dispersibility in aqueous media.

In one of the recent studies, experimental studies have shown that the adsorption capacity of HCPs is increased as the loading of $-\text{SO}_3\text{H}$ groups increases in the polymer while inversely proportional to the cross-linking density of the parent.²³⁹ Furthermore, chemisorption was found to be the main adsorption mechanism for SHCPs in removing pollutants *via* ionic interactions. Interestingly, SHCP1 showed around four-fold enhanced adsorption capacity for CIP compared to Norit (the commercial activated carbon) and Amberlyst-15. In another report, a 3D hyper-crosslinked porphyrin based POP (TP-PPOPs) was prepared using tetraphenylporphyrin and triptycene as monomers and the resulting solid was reacted with ClSO₃H to obtain TP-PPOPs-SO₃H, a sulfonated material with



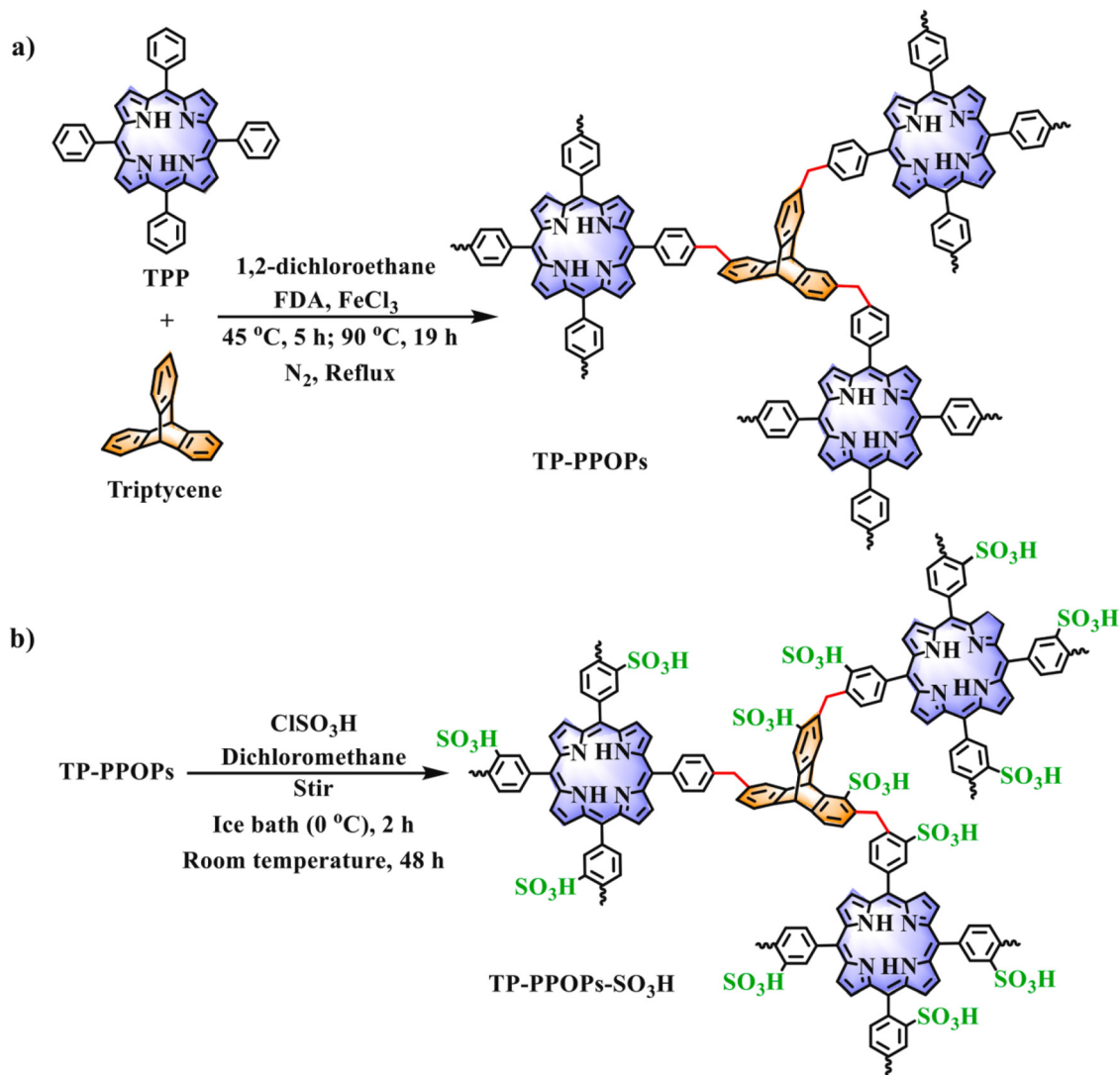


Fig. 19 Synthesis of a) TP-PPOPs and b) TP-PPOPs-SO₃H. Reproduced with permission from ref. 240 Copyright 2024 Elsevier.

enhanced electronegativity and hydrophilicity (Fig. 19).²⁴⁰ The performance of TP-PPOPs-SO₃H was studied in the adsorption of MB and CIP, achieving 1283.33 mg g⁻¹ and 485.39 mg g⁻¹ for MB and CIP, respectively. This enhanced performance was due to several factors that include strong electrostatic interactions and high surface area with extended π -conjugated structure.

A -SO₃H-decorated COF (ImCOF-SO₃H; Im stands for imine-linked) was prepared and later, converted to a COF (AmCOF-SO₃H; Am stands for amide-linked) through the reaction with Emim BF₄ (Fig. 20)²⁴¹ and its performance was studied in the adsorption of enrofloxacin (ENR), enoxacin (ENO), and norfloxacin (NOR). The adsorption capacities were 614 mg g⁻¹, 615 mg g⁻¹, and 579 mg g⁻¹, for ENR, ENO and NOR, respectively, representing a 2.5-fold enhancement over ImCOF-SO₃H. This enhanced activity of AmCOF-SO₃H is due to the stronger electrostatic interactions with -SO₃H groups and the facile formation of hydrogen bonding with the amide groups. These interactions are further complemented by π - π stacking and favorable pore-size matching effects. Notably, AmCOF-SO₃H showed 90% removal of

fluoroquinolone antibiotics, thus demonstrating the practical application of this solid in wastewater treatment.

SHCPs were synthesized and their performance was tested for the adsorptive removal of various antibiotic pollutants under environmentally relevant conditions.¹⁶⁶ The SHCPs were capable of highly efficient removal of antibiotic pollutants at relatively high (30 mg L⁻¹) and low (50 μ g L⁻¹) initial concentrations, both from a simple as well as complex water matrices. Interestingly, the rate of CIP removal and the adsorption capacity observed for SHCP was \sim 758 mg g⁻¹ which is approximately two-fold than that of both post-synthetically sulfonated polymers (\sim 477 mg g⁻¹) and Amberlyst-15 (\sim 439 mg g⁻¹). These comparisons clearly indicate the beneficial effects of the highly functionalized SHCP as a potential adsorbent for the selective removal of organic pollutants from aqueous media.

4.3. Metal ion capture and recovery

To enable the selective capture and recovery of metal ions, -SO₃H groups are integrated into PPOPs, creating powerful



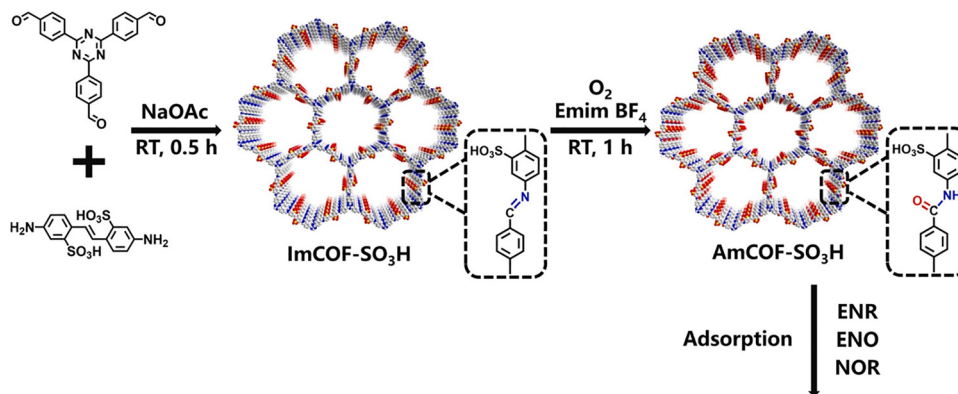
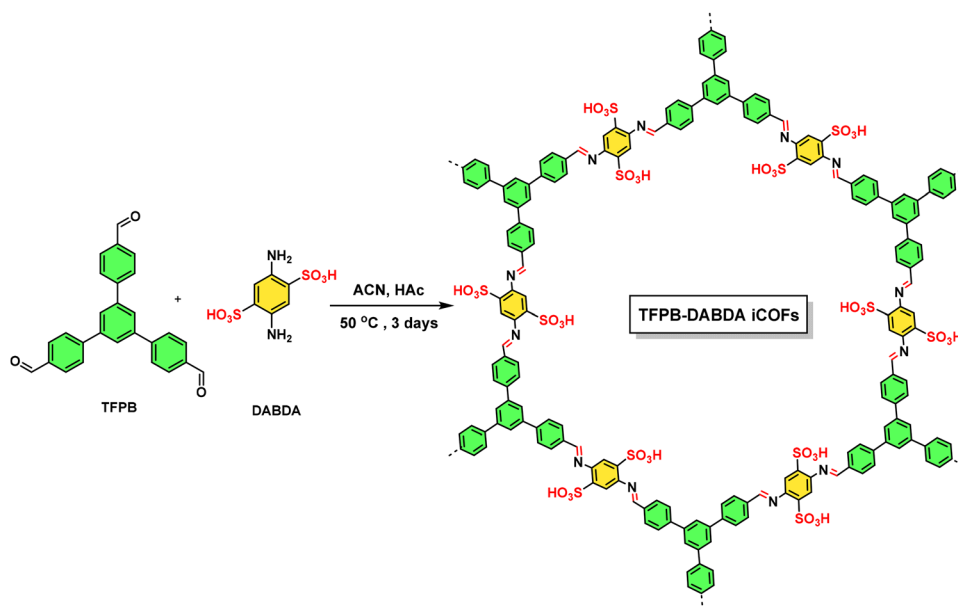


Fig. 20 Synthesis of AmCOF-SO₃H catalyst. Reproduced with permission from ref. 241 Copyright 2025 Elsevier.

chelating platforms. This functionality is particularly transformative for the challenging separation of lanthanides, whose nearly identical chemical properties render conventional separation inefficient. SPOPs overcome this by exhibiting unique and tunable ion-selectivity, often based on ionic radii. Lanthanides are a group of important elements that have been widely used in many fields.²⁶² Their separation is crucial for technical applications, but it remains a troublesome task due to their subtly different properties. In one of the recent studies, a nanoporous sulfonic ionic COFs (iCOFs) was obtained by reacting 1,3,5-tris(*p*-formylphenyl)benzene (TFPB) and 2,5-diaminobenzene-1,4-disulfonic acid (DABDA) to obtain TFPB-DABDA iCOFs and its performance was assessed in the selective adsorption separation of lanthanide elements (Scheme 10).²⁴² The experimental results indicate that the adsorption capacity of TFPB-DABDA iCOFs is highly sensitive to the ionic radius of the lanthanides. The adsorption capacity for light rare earth elements (LREEs: La³⁺, Ce³⁺, Pr³⁺, Nd³⁺, Sm³⁺, Eu³⁺, and Gd³⁺) gradually increased,

while for the trend for heavy rare earth elements (HREEs: Tb³⁺, Dy³⁺, Ho³⁺, Er³⁺, Tm³⁺, and Lu³⁺) was different, initially increasing at lower concentrations and then decreasing significantly with further concentration increases. Furthermore, TFPB-DABDA iCOFs exhibited the maximum adsorption capacity of 0.1554 mmol g⁻¹ for early lanthanides, while for late lanthanides it is negligible. The separation mechanism was due to the uniform distribution of -SO₃H groups attached within the ordered channels of the COFs.

A -SO₃H-functionalized HCPs (SHCPs) was synthesized with 3.27 mmol g⁻¹ density of -SO₃H having 1044.6 m² g⁻¹ specific surface area with uniform micropores, and superhydrophilicity.²⁴³ These inherent properties enabled fast adsorption kinetics (30 min) and high adsorption capacities for REEs (106.78 mg g⁻¹ for La, 111.99 mg g⁻¹ for Eu, and 126.27 mg g⁻¹ for Lu). Furthermore, the material showed extremely high REE³⁺/Al³⁺ selectivity in a sulfate medium (SF(La/Al) = 6929, SF(Eu/Al) = 4810, SF(Lu/Al) = 3003), surpassing most existing REE adsorbents.



Scheme 10 Synthesis of sulfonated TFPB-DABDA iCOFs. Adapted from ref. 242.



When applied to wastewater, SHCP adsorbed 87% of the total REE³⁺, while the adsorption rate for Al³⁺ was zero. These results imply that the high selectivity stems from the difference in binding stability between REE³⁺ and Al³⁺ complexes with sulfate ions, demonstrating that -SO₃H groups offer a promising approach for REE recovery from sulphate-rich aqueous solutions. Beyond REEs, SPOPs have also been applied to the detection and removal of other critical and toxic species. The Schiff base condensation of 2,4,6-triformylphloroglucinol (Tp) and 3,7-diaminobenzothiazole (DAS) resulted in an imine-based COF (TAS-COF).²⁴⁴ The benzothiophene sulfone unit in TAS-COF endows oxidase-like activity upon exposure to visible light irradiation by generating superoxide radicals. Thus, under visible light irradiation, TAS-COF photocatalyzes the oxidation of colorless 3,3',5,5'-tetramethylbenzidine (TMB) to blue oxTMB. The subsequent coordination of uranyl ions (UO₂²⁺) with oxTMB quenches this blue color, providing the basis for a facile colorimetric sensor. Using this strategy, TAS-COF achieved a detection limit of 0.07 μmol L⁻¹ for UO₂²⁺. A sulfonated COF, designated TFPOTDB-SO₃H, was synthesized by reacting DABSA and 2,4,6-tris-(4-formylphenoxy)-1,3,5-triazine, and its efficacy was tested for the adsorption of As(III) (Fig. 21).²⁴⁵ The material achieved a 97.1% removal rate of As(III) within 10 min, corresponding to a high adsorption uptake of 345 mg g⁻¹ at pH 8 and room temperature. Furthermore, the influence of other competing anions was studied and observed a reduction of As(III) uptake by around 1–7%. Moreover, the adsorbent exhibited excellent reusability, retaining 89% of its initial uptake capacity after four successive adsorption-desorption cycles.

SPOPs also show versatility in adsorbing alkali metals. A -SO₃H-functionalized (SHCP) was synthesized through FC reaction and the performance of SHCP was studied in the adsorption of Rb⁺ and Cs⁺.²⁴⁶ The adsorption efficiencies for Rb⁺ and Cs⁺ using SHCP was 94.7% and 95.9% at 30 min. Interestingly, SHCP showed a good adsorption selectivity for Rb⁺ and Cs⁺ ions in the

presence of several competitive ions. The physical doping and freeze-drying of sulfonated CMP (SCMP) with carboxymethyl cellulose (CMC) resulted in a composite aerogel (CMC/SCMP) and its activity was tested in the adsorption of Pb²⁺ and MB.²⁴⁷ The adsorption capacity was up to 94.93 and 294.84 mg g⁻¹, respectively for Pb²⁺ and MB. The characterization and experimental data indicate that this activity is attributed to chelation, electrostatic attraction, H-bonding and π-π interaction.

4.4. Gas separation

In gas separation applications, the integration of -SO₃H groups endows porous polymers with high selectivity for gases like CO₂ and NH₃. These strongly acidic, polar sites chemisorb basic gas molecules, significantly boosting their uptake over non-polar species such as N₂ and CH₄. This is evidenced by dramatically increased isosteric heats of adsorption and exceptional selectivity, with reported CO₂/N₂ values exceeding 400 for some sulfonated networks. Furthermore, when integrated into membranes, sulfonate groups induce tighter polymer chain packing, which critically enhances diffusion selectivity. This synergistic effect between improved sorption selectivity (from strong gas-functional group interactions) and enhanced diffusion selectivity (from modified polymer morphology) leads to a superior overall separation performance that often surpasses the upper bounds of current trade-off curves for key gas pairs like CO₂/N₂ and O₂/N₂. For instance, a study on HCPs with PIMs demonstrated how functionalizing the polymer backbone with polar groups like -NO₂, -NH₂, and -HSO₃ tunes their selectivity for CO₂ over N₂ and CH₄.¹³⁶ Although the introduction of these functional groups generally reduced the overall porosity, this was more than compensated for by a significant improvement in CO₂ affinity. The sulfonated polymers achieved the highest CO₂ uptake capacity of 298 mg g⁻¹ (6.77 mmol g⁻¹). In terms of selectivity, the aminated (-NH₂) polymers performed best, reaching values of 26.5 for CO₂/N₂ and up to 8.6 for CO₂/CH₄.

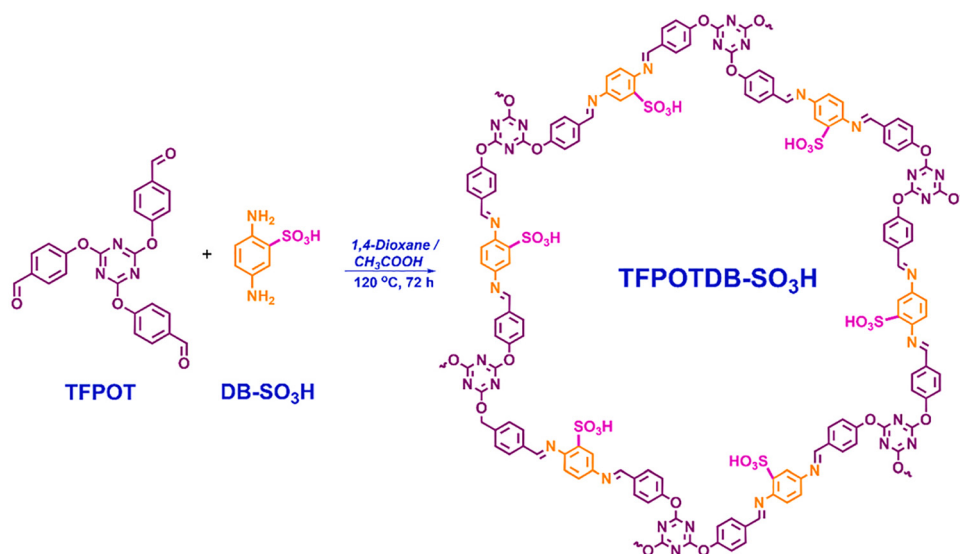


Fig. 21 Synthesis of TFPOTDB-SO₃H solid catalyst. Reproduced with permission from ref. 245 Copyright 2024 Elsevier.



The superior performance of these functionalized PIMs is attributed to a synergistic contribution from the residual porosity, the specific nature of the functional group, and an optimal isosteric heat of adsorption.

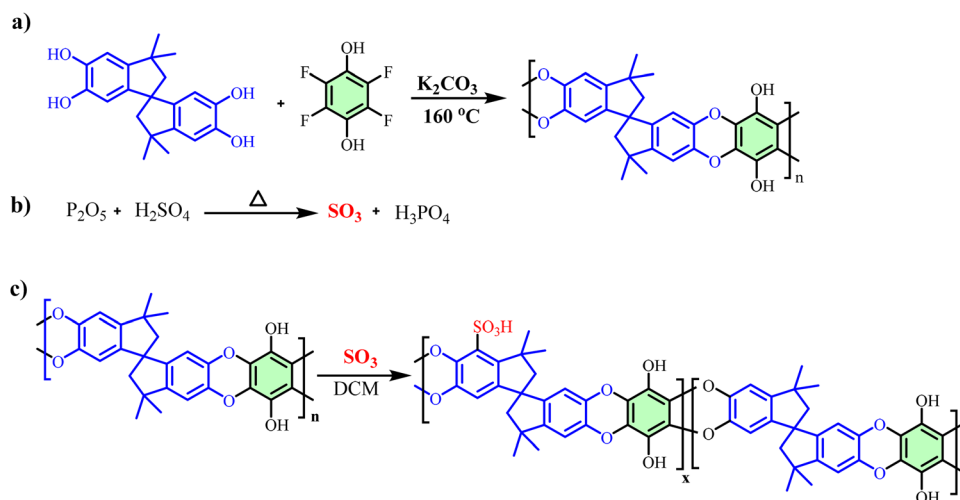
A COF (TpPa-SO₃H) was prepared *via* a modified mechanical grinding synthetic method and further functionalized with metal sites by post-loading to obtain TpPa-SO₃Cu_{0.5}.²⁴⁸ Due to the presence of abundantly accessible active sites and well-ordered COF channels, TpPaSO₃Cu_{0.5} demonstrated superior adsorption kinetics and a high capacity of 30.45 mg N g⁻¹, outperforming most reported adsorbents (<0.001–0.994 g mg⁻¹ min⁻¹ and 0–25 mg N g⁻¹). Besides, TpPa-SO₃Cu_{0.5} showed a selectivity coefficient 328 times higher than to TpPa-SO₃H in real water (10 mg N L⁻¹, pH = 10). This solid exhibited a high NH₃ recycling ratio (~95%) during five adsorption-regeneration cycles. In a separate approach, a hyper-crosslinked POP was sequentially oxidized and sulfonated in a post-synthetic treatment to generate a porous solid decorated with readily accessible carboxylic and -SO₃H groups.²⁴⁹ Interestingly, the polymer was coated with poly(dimethylsiloxane) and the resulting solid exhibited a 40-times increase (from 0.04 to 1.41 mmol g⁻¹) compared to non-modified polymer towards a low-pressure NH₃ adsorption; key merits of this approach include easy preparation, cost-effectiveness, and scalability for mass production. In another study, sulfonated polyimide (SPI) foams with tunable surface wettability were developed for methane hydrate formation.²⁵⁰ The research demonstrated that an SPI foam with a balanced hydrophobic-hydrophilic surface significantly enhanced formation kinetics. Under static conditions, the optimized foam achieved an extremely short induction period and a rapid hydrate formation rate of 2.3 V V⁻¹ min⁻¹, along with a high storage capacity of 119 V V⁻¹, all without the use of chemical additives. An early investigation reported that sulfonated polymeric networks (PPN-6-SO₃H and its Li⁺ salt, PPN-6-SO₃Li) exhibited a significant enhancement in both the isosteric heat of CO₂ adsorption and the overall CO₂ uptake capacity.⁵³ Ideal

adsorption solution theory calculations using single-component-isotherm data and a 15/85 CO₂/N₂ ratio at 295 K and 1 bar, revealed exceptionally high selectivity for CO₂ over N₂ (155 and 414 for PPN-6-SO₃H and PPN-6-SO₃Li, respectively). Three SHCPs with distinct structural features were synthesized and their performance was evaluated in NH₃ adsorption.²⁵¹ Among the various solids screened, SHCP-1 showed superior NH₃ uptake of 11.54 mmol g⁻¹ at 30 °C and 1 bar. Interestingly, NH₃ adsorption capacity was more than 10 mmol g⁻¹ even after ten regeneration cycles, thus illustrating its stability. This superior performance of this solid is attributed to the presence of well-developed micro/mesoporous structure consisting of abundant adsorption sites which favors the chemisorption between -SO₃H groups and -NH₃. In another study, a PIM-1 membrane was modified through straightforward sulfonation with a sulfur trioxide to afford sulfonated PIM-1 (SPIM-1) membranes (Scheme 11).¹³⁵ Characterization data revealed the attachment and uniform distribution of -SO₃H groups along the polymer backbone. Further, SPIM-1-6 (6 stands for PIM-1 was sulfonated for 6 min) showed the highest gas separation performance, with selectivity values approaching or exceeding the latest trade-off limits for O₂/N₂, CO₂/N₂, H₂/N₂, and CO₂/CH₄ pairs. The superior activity can be attributed to the SO₃H group, inducing compact polymer chain packing, which remarkably enhances diffusion selectivity. After 60 days of aging, the SPIM-1-6 membrane exhibited even higher selectivity (H₂/N₂ = 125, O₂/N₂ = 8.43) alongside high gas permeabilities (H₂ = 1077 Barrer, O₂ = 73.4 Barrer).

5. Energy-related applications

5.1. Proton conductivity

Proton-conducting materials are central to electrochemical energy technologies because they govern the efficiency, durability, and cost of fuel cells, sensors, and chemical reactors.^{263,264} Among them, polymer electrolyte membrane fuel cells (PEMFCs)



Scheme 11 Synthesis of a) PIM-1; b) the SO₃ sulfonation solution and c) SPIM-1 membranes, where x and y vary with sulfonation time. Adapted from ref. 135.



are particularly attractive for portable, automotive, and stationary applications. In such devices, the polymer electrolyte membrane is widely regarded as the “heart” of the system, since its properties dictate key performance metrics including power density, operational lifetime, and system complexity. To operate reliably, a PEM must simultaneously provide high proton conductivity, negligible electronic conductivity, minimal fuel and oxidant crossover, and robust thermal, chemical, and mechanical stability, while remaining cost-effective and processable at scale.^{265,266} These stringent and often competing requirements have driven intensive research not only into dense fluorinated and hydrocarbon polymers, but increasingly into architecturally tunable porous frameworks that allow decoupling and rational optimization of ion transport, mechanical integrity, and chemical robustness.^{267,268}

Historically, organic proton-conducting polymers have dominated the PEM field. Nafion[®], a perfluorosulfonic acid (PFSA) membrane developed by DuPont, has set the technological benchmark owing to its high proton conductivity under humidified conditions, well-developed morphology, and proven long-term operation in devices. Together with Nafion[®], high-temperature polybenzimidazole (PBI) membranes and sulfonated hydrocarbon polymers such as sulfonated poly(ether ether ketone) (sPEEK) represent the state-of-the-art in commercial and pre-commercial systems.^{269,270} Despite these successes, however, conventional polymeric membranes exhibit persistent limitations. Nafion[®] suffers from a sharp loss of conductivity at elevated temperatures and reduced relative humidity, complex and energy-intensive processing, and concerns related to fluorinated waste and cost. Likewise, phosphoric-acid-doped PBI and sulfonated polyaryl membranes often display trade-offs between proton conductivity, oxidative and mechanical stability, and dimensional control, as highlighted in recent benchmark studies.^{271,272} Collectively, these issues indicate that even optimized PFSA and hydrocarbon membranes do not fully satisfy the simultaneous requirements of high conductivity, durability, and sustainability, motivating the exploration of alternative architectures such as sulfonated POPs.

To address the persistent trade-offs inherent to conventional polymer electrolytes, research has increasingly turned to composite and hybrid membrane architectures that introduce auxiliary phases capable of reorganizing local proton-transport environments. Representative approaches include acid-base complexes such as PBI-H₃PO₄,²⁷³ heterocycle-assisted proton conductors,²⁷⁴ and nanostructured Nafion[®]-CNT hybrids,²⁷⁵ as well as inorganic-organic systems such as CsHSO₄-SiO₂ and CsH₂PO₄-SiO₂.²⁷⁶ While these strategies demonstrate that hydrogen-bond organization, water retention, and phase stability can be modulated through secondary components, their impact remains fundamentally constrained by the limited architectural freedom of dense polymer matrices, which restricts long-range transport pathways and hinder deliberate positioning of acid sites. Consequently, improvements tend to be incremental rather than transformative, reinforcing the need for frameworks in which porosity, acidity, and transport channels can be co-engineered from the molecular level. Architecturally tunable porous materials particularly SPOPs

and SCOFs²⁶⁷ directly address this gap by enabling decoupled control over proton-donating groups, pore topology, and conduction mechanisms, thereby offering a fundamentally different design space than conventional composite membranes.

A clear understanding of proton-transport mechanisms is essential for designing advanced electrolytes, particularly in systems where acidity, pore topology, and local chemical environments can be engineered. Proton motion in polymeric and porous materials typically arises from a combination of Grotthuss-type structural diffusion and vehicular transport by protonated species. The former operates through rapid hydrogen-bond reorganization and is generally associated with lower activation energies (<0.4 eV), while the latter involves the diffusion of hydronium or other protonated carriers and dominates when mobility is restricted or hydration is limited.²⁷⁷ In architecturally tunable porous frameworks such as SPOPs and SCOFs, these mechanisms are strongly influenced by the spatial distribution of -SO₃H groups, pore confinement, and the degree of structural ordering (Fig. 22). Periodically aligned acid sites and narrow channels can promote directional hopping, whereas disordered or highly hydrated domains favor mixed pathways. Such transitions are typically reflected in activation-energy trends obtained from impedance spectroscopy.^{278,279} Importantly, the ability to decouple pore geometry, acid-site placement, and hydration environments in porous frameworks, an option not accessible in dense polymers, creates a distinct design platform for tailoring proton-conduction pathways across a broad range of operating conditions.

Beyond classical polymers and composite membranes, crystalline and semi-crystalline porous materials have emerged as a powerful platform for advancing proton-conducting electrolytes. Among these, COFs have attracted particular attention due to their long-range structural order, well-defined pore channels, and precise chemical tunability. Constructed from lightweight elements linked through strong covalent bonds, COFs offer an unprecedented level of control over pore size, topology, and functional-group placement, enabling direct correlation between framework architecture and proton-transport behavior. In contrast to amorphous polymer systems, the periodicity of COFs allows systematic investigation of how channel confinement, acid-site alignment, and hydrogen-bond organization govern conduction mechanisms. Early studies demonstrated that COFs functionalized with acidic groups can sustain continuous proton-hopping pathways within ordered nanochannels, providing model systems to decouple structural effects from compositional complexity. Although their absolute conductivities initially lagged behind benchmark polymer membranes, these materials established critical structure-mechanism relationships that now underpin the rational design of high-performance SCOFs and related porous organic frameworks.²⁸⁰

Parallel to COFs, other POPs have emerged as a complementary and highly versatile class of proton-conducting materials. Unlike crystalline COFs, POPs are typically amorphous or semi-ordered, yet they offer exceptional chemical and thermal stability, broad synthetic flexibility, and scalable preparation routes. Their inherently disordered nature allows dense incorporation of functional groups without the strict topological



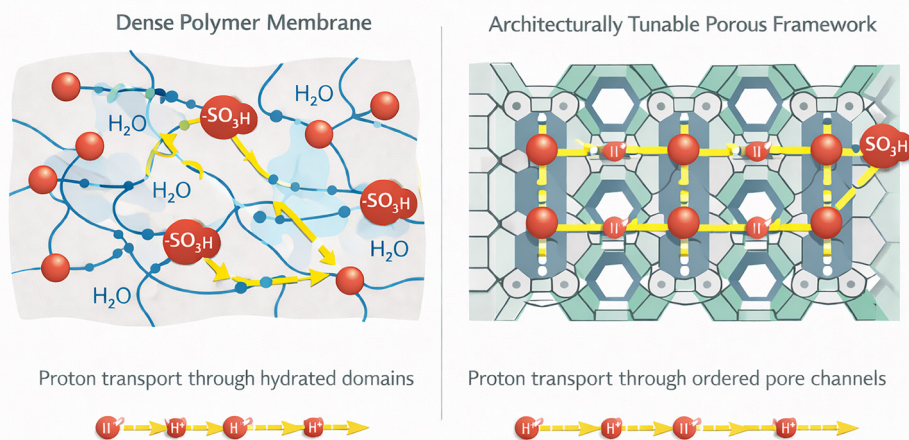


Fig. 22 Conceptual schematic of proton transport in dense polymer membranes versus architecturally tunable porous frameworks, highlighting hydration-dependent tortuous pathways and ordered pore-confined proton conduction, respectively.

constraints imposed by crystallinity. As a result, POPs provide a unique platform in which high surface area, tunable pore environments, and chemical robustness can be combined to tailor proton-transport behavior. Importantly, the absence of long-range order does not preclude efficient proton conduction; instead, localized hydrogen-bond networks formed within interconnected micropores can sustain effective transport pathways when appropriately functionalized. Compared with COFs, POPs therefore offer greater tolerance toward high acid densities and structural heterogeneity, making them particularly attractive for aggressive functionalization strategies such as sulfonation. These features position POPs as a practical bridge between model crystalline frameworks and application-oriented polymer electrolytes, especially in contexts where processability, durability, and chemical stability are as critical as absolute conductivity.²⁸¹

Among the functionalization strategies explored for porous organic frameworks, sulfonation has emerged as the most effective route to impart strong Brønsted acidity and enable efficient proton conduction (Table 5). The introduction of $-\text{SO}_3\text{H}$ groups provides abundant proton-donating sites while simultaneously promoting the formation of extended hydrogen-bond networks that favor Grotthuss-type transport. In POPs, sulfonation can be achieved either through PSM of preformed backbones or *via* the direct incorporation of sulfonated monomers during polymerization. These two approaches offer fundamentally different levels of control over acid-site distribution, framework integrity, and pore accessibility. Post-sulfonation enables high acid densities and broad applicability across diverse POP chemistries but may introduce structural heterogeneity or partial pore blockage if not carefully controlled. In contrast, in-monomer sulfonation allows more uniform placement of $-\text{SO}_3\text{H}$ groups and preserves pore connectivity, albeit often at the expense of synthetic complexity. Crucially, studies across multiple POP families consistently demonstrate that proton conductivity is governed not only by the total acid content but also by the spatial organization of

sulfonic sites relative to pore architecture and hydration domains. This decoupling of acidity, porosity, and framework stability represents a key advantage of SPOPs over dense polymer electrolytes and underpins their emerging role as tunable proton-conducting membranes.

Looking forward, SPOPs and SCOFs offer a highly promising platform for next-generation proton-conducting membranes, owing to their modular architectures and the ability to systematically tune chemical composition, pore environment, and functional-group density. These attributes enable fine control over proton-transport mechanisms that are difficult to achieve in conventional dense polymers. At the same time, key challenges remain, including achieving high conductivity under anhydrous or low-humidity conditions, ensuring long-term operational stability, and developing scalable synthesis routes. Addressing these issues is essential for translating laboratory-scale performance into practical PEMFC technologies. In the following sections, we therefore highlight representative advances in SPOPs, SCOFs, and their hybrid membranes with conventional polymers, with particular emphasis on synthetic strategies, structure–property relationships, conductivity benchmarks, and mechanistic design principles relevant to proton transport.

The development of proton-conducting POPs has predominantly relied on post-sulfonation as a versatile and chemically robust strategy to introduce strong Brønsted acid sites into stable porous backbones. Early studies on phenolic and aromatic POPs demonstrated that carefully designed frameworks can withstand harsh sulfonation conditions while preserving structural integrity, thereby enabling exceptionally high densities of covalently anchored $-\text{SO}_3\text{H}$ groups within accessible pore environments. Importantly, these systems revealed that proton conductivity is not dictated solely by acid loading; instead, the continuity of hydrogen-bond networks formed within interconnected pores plays a decisive role in facilitating efficient Grotthuss-type transport under humidified conditions.²⁸² This principle was further validated in CMPs, where rigid π -conjugated backbones provided





Table 5 Key characteristics and performance of SPOPs for energy applications

POP	BBS/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
NUS-10(R)	1,3,5-triformylphloroglucinol (TFP), DABDA	Mechanosynthesis (LAG) → Recrystallization (<i>de novo</i>)	–, 69/0.84	$3.96 \times 10^{-2}/0.21$ eV	25 °C, 97% RH	5.20 mmol g ⁻¹ (IEC)	Time-dependent conductivity (15 days, stable)	–	124
MPOPS	Cyanuric chloride and DABSA	Extended condensation polymerization (<i>de novo</i>)	114/1.27	$3.07 \times 10^{-2}/$ 0.407	80 °C, humid conditions	1.836 mmol g ⁻¹ (IEC)	Stability test (1 vs t, >16 000 s), MEA test	–	125
SBO-CMP	1,4-Phenylenediboronic acid, 4,4'-biphenyldiboronic acid, tetrabromobisolefin	Suzuki–Miyaura coupling	440, 40/2	$1.29 \times 10^{-2}/0.32$	70 °C, 100% RH	27.6 wt%	–	–	126
Phloroglucinol-based POP	Phloroglucinol and terephthalaldehyde	Microwave-assisted <i>de novo</i> synthesis	992, 29/–	$7.72 \times 10^{-2}/0.34$	80 °C, 90% RH	–	Time-dependent conductivity	–	282
TaPOP-SO ₃ H	Tetrakis(4-azidophenyl)methane, 1,3,5-triethylbenzene, propargyl sulfonate	<i>In situ</i> sulfonation (click)	629.5, 63.8/1.45	$0.92 \times 10^{-2}/0.25$	80 °C, 98% RH	–	Long-life reusability test (> 20 h)	–	283
S-POP-BP-TPOT	1,4-Bis(chloromethyl)benzene (BP), 2,4,6-triphenoxy-1,3,5-triazine (TPOT)	FC alkylation followed by postsulfonation	395, 93/0.62, 0.27 Pore volume cm ³ g ⁻¹	$1.5 \times 10^{-2}/0.19$	25 °C, 95% RH	30 wt%/ 3.6 mmol g ⁻¹ (IEC)	Time-dependent proton conductivity, Fenton's reagent test, fuel-cell test	–	127
S-POP-TPM	Tetraphenylmethane (TPM), formaldehyde dimethyl acetal (FDA)	FC alkylation followed by postsulfonation	1312, 720/0.54	$0.1 \times 10^{-1}/0.23$	80 °C, 95% RH	36 wt%	Time-dependent proton conduction (6 h)	–	284
2-MeL40-S	Biphenyldicarboxaldehyde, 5-methylresorcinol (MeL), phloroglucinol (PG)	Microwave-assisted condensation followed by postsulfonation	Decreased (after MeL/ sulfonation)/–	$2.39 \times 10^{-1}/$ 0.085	80 °C, 95% RH	IEC 5.85 mmol g ⁻¹ , DoS 116.8% for 1-MeL40-S	Cyclic stability, time-dependent proton conductivity	–	285
SPAF	Tetrakis(4-bromophenyl)methane (BTMPM), Benzene-1,4-diboronic acid (BDB)	Suzuki coupling followed by gas-phase sulfonation (PSM)	1692, 124/~1	1.4/0.36 (35 kJ mol ⁻¹)	80 °C, 100% RH	5.67 mequiv g ⁻¹ (IEC)	NMR, IR, Titration (Long-term stability), ESEM	–	286
SMPAF-303	PAF-303 (BTMPM + BDB), Formaldehyde, HCl/Na ₂ SO ₃	PSM (Chloromethylation → Sulfonation)	1317(AP), 743(AT)/0.681, 0.337 pore volume cm ³ g ⁻¹	0.1/0.23 (22.6 kJ mol ⁻¹)	77 °C, 100% RH	1.73 mequiv g ⁻¹ (IEC)	High thermal stability (500 K)	–	287
50 wt% SPEEK@Ox-BD-COF-SO ₃ H	Tri(4-aminophenyl)benzene (TAPB), benzo[d][1,3]dioxole-4,7-dicarbaldehyde, SPEEK	Sequential PSM (oxidation, sulfonation) followed by infiltration	2013, 493/2.1, 1.7	$6.22 \times 10^{-3}/$ 0.199	90 °C, 98% RH	COF: 6.2 wt% SO ₃ H	Repeated conductivity tests (48-h run)	–	135
PyTA-DHTA-COF-SO ₃ H	4,4',4''-(pyrene-1,3,6,8-tetrayl)tetraamine (PyTTA), 2,5-dihydroxyterephthalaldehyde (DHTA)	Solvothermal condensation followed by PSM	2013, 493/2.1, 1.7	$2.0 \times 10^{-2}/0.38$	70 °C, 100% RH	19.5 wt%	Time-dependent proton conduction (8 h, stable)	–	283
S-COF-2	2,4,6-Trimethoxybenzene-1,3,5-tricarbaldehyde, DABDA	<i>De novo</i> synthesis (solvothermal)	351, 139/1.30, 1.26	$1.5 \times 10^{-2}/0.17$	25 °C, 95% RH	15.88 wt%/ 40.2% content	Time-dependent proton conduction (24 h, stable)	–	137
H ₃ PO ₄ @COF-SO ₃ H	1,3,5-tris(<i>p</i> -formylphenyl)benzene, DABDA	Solvothermal synthesis, mechanochemical impregnation (PSM)	53.51, 1.74/3.51	$3.57 \times 10^{-2}/$ 0.14 eV	Ambient conditions	–	Proton battery test, cycling stability (1000 cycles)	–	288
SPEEK/COF-EN ₃	Hexahydroxybenzene, tetrafluoroterephthalonitrile	Solvothermal synthesis	–/–	$2.17 \times 10^{-1}/$ 0.19 eV	80 °C, 90% RH	Speek 77%	Cell performance	–	289
COF-Fx-SO ₃	2,4,6-tris(4-formylphenyl)pyrroline, DABSA	Solvothermal synthesis	233/1.7	$1.93 \times 10^{-2}/$ 0.24 eV	140 °C, anhydrous conditions	–	Time-dependent conductivity	–	290
				$7.7 \times 10^{-3}/$ –	90 °C, 97% RH				



Table 5 (continued)

POP	BBS/monomers	Functionalization	BET surface area (m ² g ⁻¹)/pore size (nm)	Application	Performance metric	Acid density (mmol g ⁻¹)	Stability assessment	Scale up (yes/no)	Ref.
PA@TpDMAP/ Pa-SO ₃ H	DABSA, dimethylaminopyridine, triformylphloroglucinol	Solvothermal followed by PSM	247, 987/2.21	1.56 × 10 ⁻² / 0.19 eV	140 °C, anhydrous conditions	Speek 83%	AC electro-chemical impedance spectroscopy	—	291
SPEEK/ HPW@COF	1,3,5-Triformylphloroglucinol, 3,5-diamino-1,2,4-triazole	Solvothermal synthesis of triazole COF into PSM	164, 92/1.44	1.32	75 °C, 100% RH	1.75 mmol g ⁻¹ (IEC)	Time-dependent conductivity	—	128
SPEEK/ TpPaSO ₃ He5	1,3,5-triformylphloroglucinol, DABSA	RT solvothermal	—/1.5 nm	3.46 × 10 ⁻¹ / 0.27 eV	80 °C, 100% RH	2.34 mmol g ⁻¹ IEC	Single fuel cell performance	—	292
SPAF/SPEEK	1,5-Cyclooctadiene, 2,2'-bipyridyl, Ni(COD) ₂	PSM sulfonation	3360, 608/—	5.88 × 10 ⁻² / 0.13 eV	RT in pure water	SPEEK sulfonation 70.6%	Fuel cell durability test	—	293
PA-PBI/SCOF	1,3,5-Triformylphloroglucinol, DABSA	<i>In situ</i> synthesized in PBI matrix.	—/—	5.42 × 10 ⁻¹ /4.46 kJ mol ⁻¹	80 °C, 100% RH	—	PA retention stability	—	294
MIP-COF/SPBI	Melamine, isophthalaldehyde	Solvothermal synthesis	437.18/pore volume 0.55 cm ³ g ⁻¹	1.08 × 10 ⁻¹ /— (~0.046 eV)	80 °C, 100% RH	2.69/3.70 meq g ⁻¹	Fuel cell testing, oxidative stability (93.14%)	—	295
SBCNF/ TpBd-2SO ₃ H	4,4'-Diamino-3,3'-biphenyldisulfonic acid, 1,3,5-triformylphloroglucinol	Interfacial synthesis	—/—	1.48 × 10 ⁻¹ /0.32	80 °C, 100% RH	3.27 mmol g ⁻¹ IEC	Time-dependent conductivity	—	296
NUS-9/ Nafion [®]	DABSA, 1,3,5-benzenetricarboxaldehyde	Interfacial synthesis	95.8/3	2.1 × 10 ⁻¹ / <0.4	95 °C, 100% RH	3.67 wt% S content 0.9–1.02 mmol g ⁻¹ IEC	Fuel cell testing	—	297
BD(SO ₃ H) ₂ / PTFE	2,4,6-Triformylphloroglucinol, 4,4'-diamino-3,3'-biphenyldisulfonic acid	Solvothermal synthesis followed by casting onto porous PTFE	—/1.78	2.49 × 10 ⁻¹ /—	80 °C, 100% RH	29.2 wt% IEC	OCV test period under a galvanostat	—	298

*The first and second values indicate measurements before and after sulfonation, respectively (if given).

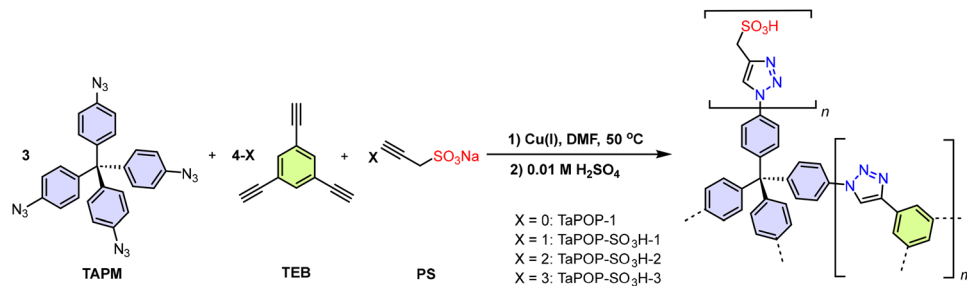


Fig. 23 Synthetic route for TaPOP and TaPOP-SO₃H structures. Adapted from ref. 283.

mechanically resilient platforms for post-sulfonation. Despite partial reductions in surface area after functionalization, SCMPs retained effective proton-transport pathways and exhibited activation energies consistent with dominant proton hopping, underscoring that pore connectivity and acid-site distribution can outweigh absolute porosity.¹²⁶ More recently, click-derived triazole-based POPs have introduced an additional level of mechanistic sophistication by combining heterocyclic nitrogen sites as intrinsic proton relays with -SO₃H groups as proton sources. The cooperative interaction between triazole units and -SO₃H functionalities promotes dense hydrogen-bond networks and enhanced water affinity, leading to conductivity enhancements of several orders of magnitude compared with pristine polymers (Fig. 23).²⁸³ Together, these representative post-sulfonated POP systems establish critical design principles grounded in framework robustness, preservation of accessible pore networks, and cooperative acid-base interactions that continue to guide the rational development of high-performance SPOP proton-conducting membranes.

Among post-sulfonated POP families, triazine-rich networks are particularly instructive because their electron-deficient nodes simultaneously facilitate electrophilic sulfonation and enhance hydrophilicity and water retention, two factors that are critical for sustaining continuous hydrogen-bond networks. In sulfonated triazine POPs prepared *via* FC polymerization, dense and uniformly distributed -SO₃H groups lead to proton conductivities approaching 10⁻² S cm⁻¹ at room temperature and high relative humidity, accompanied by low activation energies (~0.19 eV), indicative of a dominant Grotthuss-type transport mechanism (Fig. 24a).¹²⁷ Importantly, comparative studies reveal that triazine-containing

frameworks consistently outperform non-heteroatom analogues at comparable sulfonation levels, underscoring the cooperative role of backbone polarity and acid density. Beyond amorphous systems, crystalline sulfonated POPs incorporating triazine motifs demonstrate that periodic alignment of -SO₃H groups within ordered micropores further lowers transport barriers and enhances humidity tolerance, reaching conductivities above 10⁻² S cm⁻¹ under both hydrous and near-anhydrous conditions (Fig. 24b).¹²⁵ These studies establish that proton transport in sulfonated triazine POPs is governed not solely by acid content, but by a synergistic interplay between framework electronics, pore architecture, and water-mediated hydrogen-bond connectivity, providing a transferable design rationale for next-generation polymer electrolytes.

Rather than indicating that higher sulfonation degrees alone guarantee superior performance, recent studies on post-sulfonated POPs collectively reveal that proton conduction is an emergent property of framework-level organization. Effective SPOPs operate in a regime where chemical functionality, pore continuity, and local polarity are co-optimized, allowing proton transport pathways to persist even under conditions where dense polymer membranes fail. In this context, post-sulfonation should be viewed not merely as a functionalization step, but as a structural perturbation that reshapes hydration domains and transport topology within the framework. Architectures that tolerate this perturbation without collapsing pore accessibility or disrupting inter-pore connectivity consistently exhibit lower activation barriers and improved humidity tolerance. Importantly, these insights reposition post-sulfonated POPs as a distinct materials class in which conductivity can be tuned independently from mechanical reinforcement and chemical stability a level of decoupling that

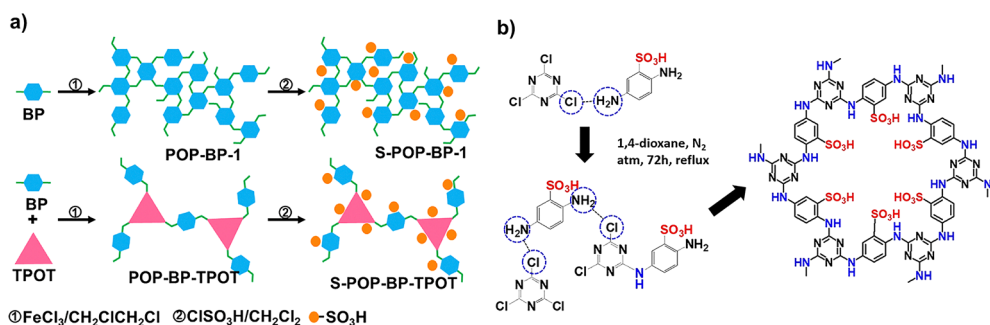


Fig. 24 a) Design strategy and synthetic pathway for triazine-derived POPs and S-POPs. ref. 127 Copyright 2020 American Chemical Society; b) depiction of the formation of MPOPS-1. Reproduced with permission from ref. 125 Copyright 2020 American Chemical Society.



remains inaccessible in conventional polymer electrolytes. This shift from composition-driven to architecture-driven optimization defines the principal conceptual advance offered by post-sulfonated POPs for proton-conducting membrane design.

Beyond framework selection and sulfonation chemistry, recent advances have increasingly targeted synthetic strategies that explicitly couple acid strength with mesoscale transport organization. Rather than relying on incremental increases in sulfonation degree, these approaches aim to synchronize proton mobility, hydration stability, and long-range pathway continuity within porous architectures. A representative strategy employed hierarchically porous SPOPs with high densities of covalently tethered $-\text{SO}_3\text{H}$ groups, achieving proton conductivities close to $10^{-1} \text{ S cm}^{-1}$ at 80°C and 95% RH while preserving mechanical and chemical integrity during prolonged operation. Notably, this work demonstrated that sustained high conductivity arises from continuous, interconnected proton-transport pathways enabled by hierarchical pore organization, rather than from acid loading alone.²⁸⁴ Building on this structural paradigm, a subsequent study introduced electronic modulation of $-\text{SO}_3\text{H}$ sites through the incorporation of electron-withdrawing substituents into the POP backbone. This electronic tuning enhanced intrinsic Brønsted acidity and stabilized

hydrogen-bond networks, resulting in record conductivities of $0.15\text{--}0.239 \text{ S cm}^{-1}$ and exceptionally low activation energies ($0.085\text{--}0.096 \text{ eV}$) at 80°C and 90% RH.²⁸⁵ Together, these studies establish that high-performance proton conduction in SPOPs emerges from the synergistic optimization of pore connectivity and acid-site electronics, providing a clear mechanistic design framework that transcends conventional structure–property trade-offs in polymer electrolytes (Fig. 25).

Alongside these developments, PAFs have emerged as a closely related but conceptually distinct subclass of POPs, in which rigid, highly connected aromatic skeletons enable exceptionally high densities of $-\text{SO}_3\text{H}$ sites without catastrophic loss of structural integrity. In contrast to more flexible polymeric backbones, the intrinsic rigidity of PAFs suppresses excessive segmental motion and dimensional swelling, allowing proton transport to be governed predominantly by pore-confined hydrogen-bond networks rather than bulk polymer relaxation. Early sulfonated aromatic frameworks demonstrated that densely packed $-\text{SO}_3\text{H}$ groups directly anchored to rigid aromatic cores can sustain ultrafast proton conduction approaching 1 S cm^{-1} under humidified conditions, a performance attributed to minimized activation barriers and continuous acid-lined micropores (Fig. 26).²⁸⁶ More recent advances introduced methylene-linked

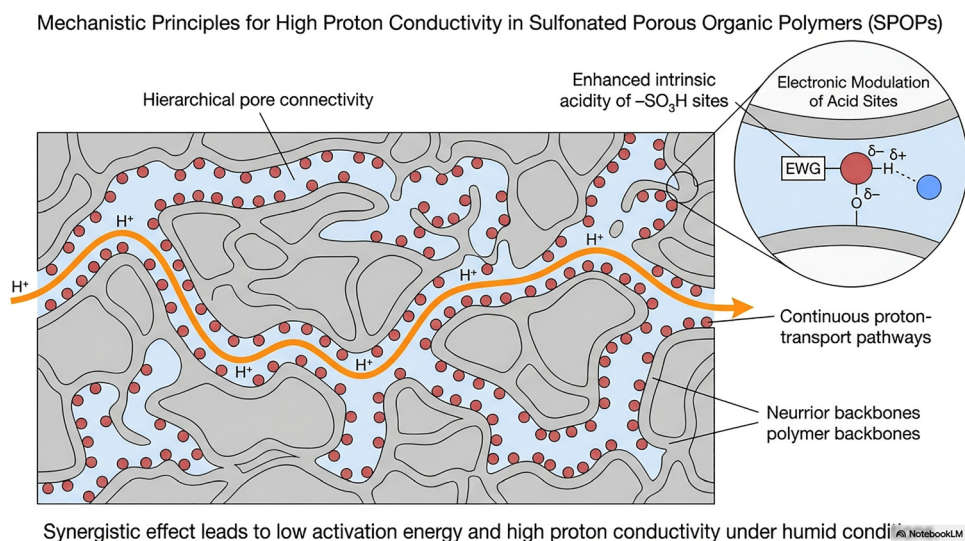


Fig. 25 Hierarchically connected hydrated pores combined with electronically tuned, covalently tethered $-\text{SO}_3\text{H}$ sites enable continuous Grotthuss-type proton transport, giving rise to high proton conductivity and low activation energy in SPOPs.

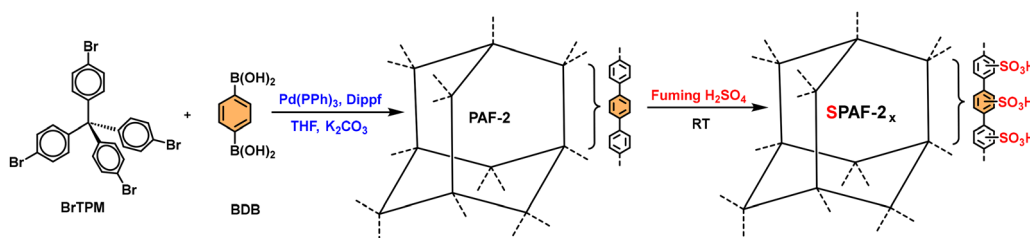


Fig. 26 Synthesis of SPAF-2x from tetrakis(4-bromophenyl)methane (BrTPM) and 1,4-diboronic acid (BDB) via PAF-2 formation and subsequent sulfonation (x = degree of sulfonation). Reproduced with permission from ref. 286 Copyright 2023 American Chemical Society.



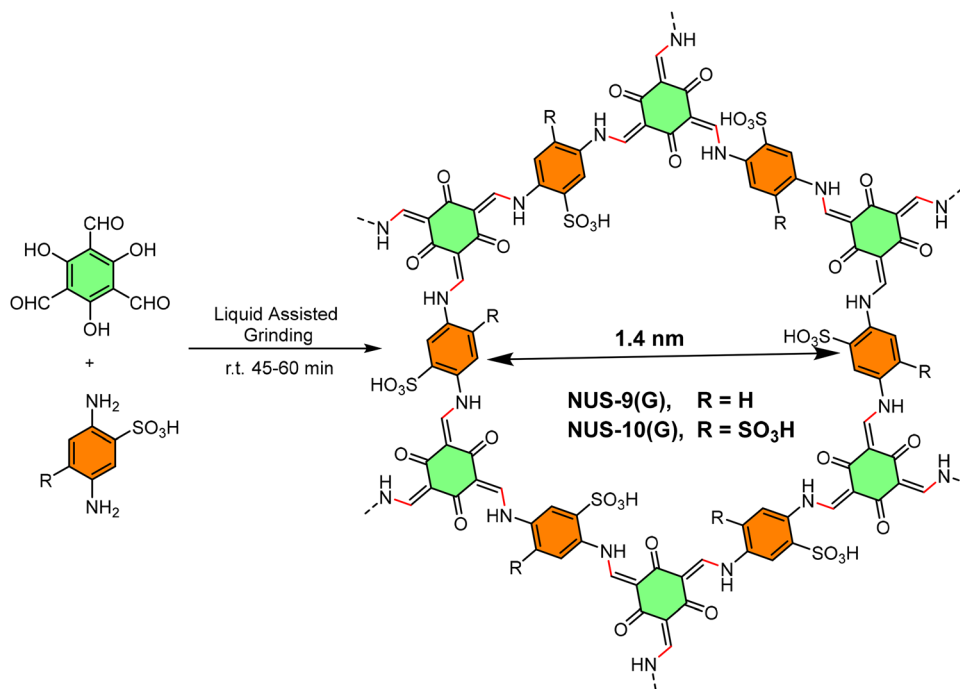


Fig. 27 Synthetic scheme of sulfonated COFs NUS-9(G) and NUS-10(G) via liquid assisted grinding at room temperature. Adapted from ref. 124.

$-SO_3H$ side chains into microporous aromatic frameworks, decoupling acid-site stability from the aromatic backbone and significantly enhancing thermal robustness. In these systems, side-chain engineering preserves strong Brønsted acidity while mitigating the thermal fragility associated with directly sulfonated aromatic rings, enabling conductivities spanning from 10^{-7} to 10^{-1} S cm^{-1} depending on hydration level. Importantly, these studies reveal that proton transport in SPAFs is highly sensitive to local polarity modulation within the pore environment: bottleneck effects dominate at low relative humidity due to electrolyte clustering, whereas fully hydrated pores eliminate transport barriers and enable near-bulk proton mobility.²⁸⁷ PAF-based systems highlight a critical design principle distinct from both COFs and conventional POPs, that rigid aromatic connectivity combined with controlled acid-site tethering can simultaneously maximize conductivity, thermal stability, and confinement, positioning SPAFs as a key bridge between amorphous SPOPs and crystalline framework electrolytes.

While SPOPs and SPAFs clearly demonstrate that high acid density and pore confinement can sustain efficient proton transport even in disordered architectures, their amorphous nature inevitably limits precise control over transport pathways and acid-site periodicity. In such systems, proton conduction is largely governed by local hydrogen-bond networks and mesoscale connectivity, which complicates the decoupling of intrinsic structure-property relationships from statistical disorder. To overcome these limitations and to gain molecular-level insight into proton-transport mechanisms, increasing attention has shifted toward COFs. Owing to their crystallinity, long-range order, and modular synthetic design, COFs enable the periodic alignment of acidic functionalities within well-defined nanochannels, providing an

unparalleled platform to systematically interrogate the roles of pore geometry, acid-site spacing, and hydrogen-bond organization in proton conduction. As a result, COFs have emerged not only as promising proton-conducting materials in their own right, but also as model systems that complement amorphous SPOPs by enabling fundamental structure-transport correlations that are otherwise difficult to access in disordered frameworks.

A decisive milestone in this context was reported by Peng and co-workers in 2016, who introduced a mechanoassisted strategy to synthesize intrinsically SCOFs with densely and periodically decorated $-SO_3H$ groups along crystalline nanochannels. The resulting frameworks (NUS-9 and NUS-10) achieved proton conductivities up to 3.96×10^{-2} S cm^{-1} under humidified conditions, accompanied by activation energies of approximately 0.20 eV, consistent with a dominant Grotthuss-type transport mechanism (Fig. 27).¹²⁴ Beyond their electrochemical performance, this work was transformative in two key respects: first, it demonstrated that strong Brønsted acid sites can be permanently integrated into COF backbones without sacrificing crystallinity or structural integrity; second, it showed that such crystalline proton conductors can be processed into PVDF-based mixed-matrix membranes with long-term operational stability. By simultaneously establishing structure-defined proton transport and membrane-level processability, this study laid the conceptual and practical foundation for scalable, functionalized COFs as next-generation proton-conducting materials.

Building on the mechanoassisted SCOF paradigm, studies reported in 2021 marked a decisive shift toward structurally precise proton conductors by refining how $-SO_3H$ groups are introduced and stabilized within crystalline frameworks. In particular, work on intrinsically sulfonated two-dimensional



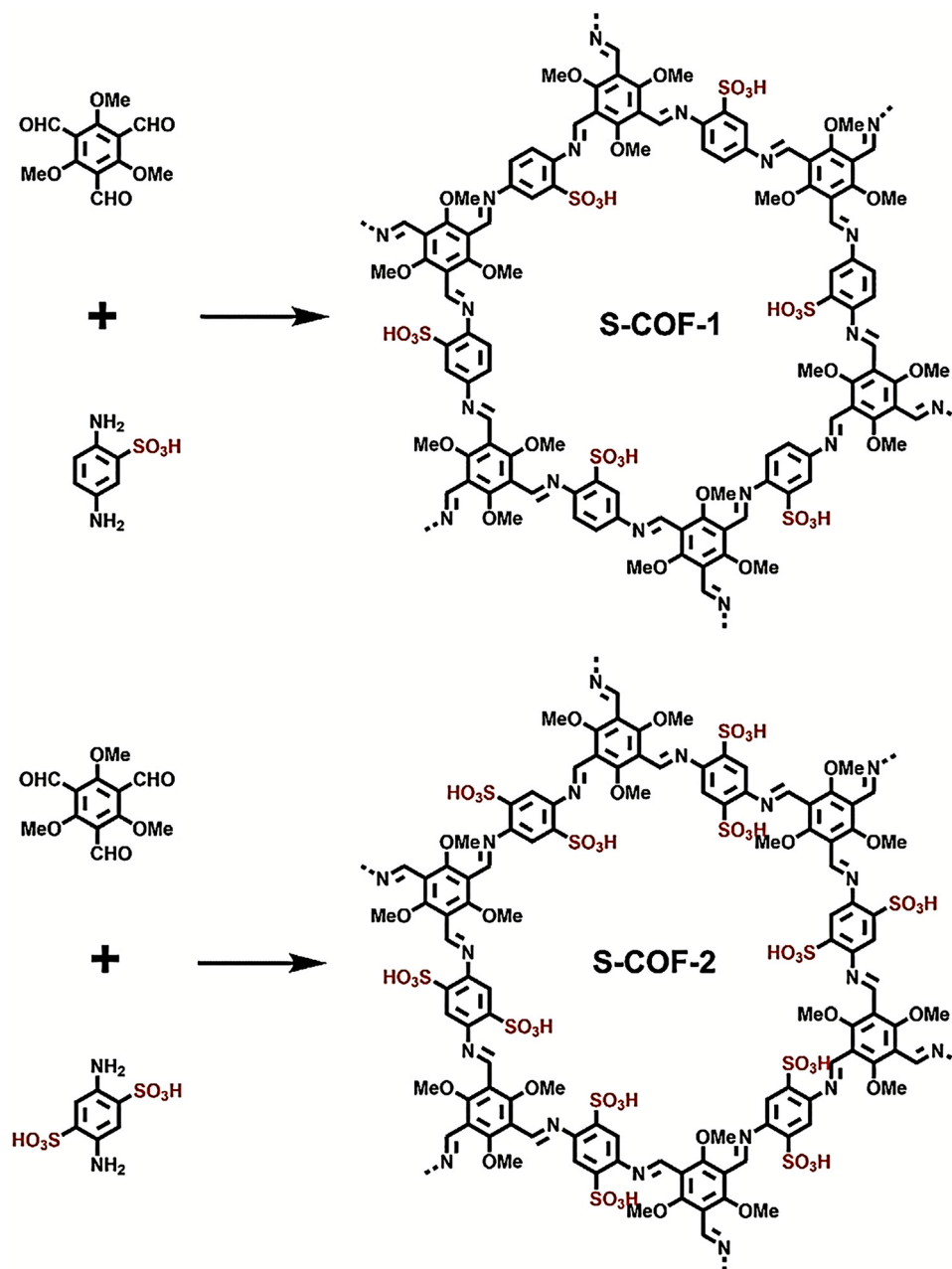


Fig. 28 Synthesis of S-COF-1 and S-COF-2 solid catalysts. Reproduced with permission from ref. 137 Copyright 2022 Wiley-VCH.

COFs demonstrated that high densities of covalently anchored $-\text{SO}_3\text{H}$ groups can be incorporated while fully preserving long-range order, enabling proton transport to be governed by periodic channel architecture rather than local disorder. These materials exhibited proton conductivities approaching $10^{-2} \text{ S cm}^{-1}$ at high relative humidity, with low activation energies ($\sim 0.17\text{--}0.20 \text{ eV}$), providing compelling evidence for dominant Grotthuss-type hopping along ordered one-dimensional nanochannels.¹⁵⁸ Complementary post-synthetic sulfonation studies further clarified the delicate balance between linker electronics and steric accessibility, showing that crystallinity retention is essential for sustaining continuous hydrogen-bond networks and minimizing transport bottlenecks.¹³³ In parallel, bottom-up approaches in which $-\text{SO}_3\text{H}$

groups were anchored directly into COF backbones eliminated acid leaching and ensured uniform proton-source distribution across channels, resulting in binder-free electrolytes with enhanced durability (Fig. 28).¹³⁷ These studies transformed SCOFs from conceptual proton conductors into structurally controlled model systems, where proton conductivity can be quantitatively correlated with acid-site periodicity, pore confinement, and framework order.

The application scope of proton-conducting COFs has recently expanded beyond classical fuel-cell membranes toward emerging proton-based energy-storage devices, where ordered ion transport and stability under demanding conditions are critical. In a notable 2025 study, a SCOF loaded with phosphoric



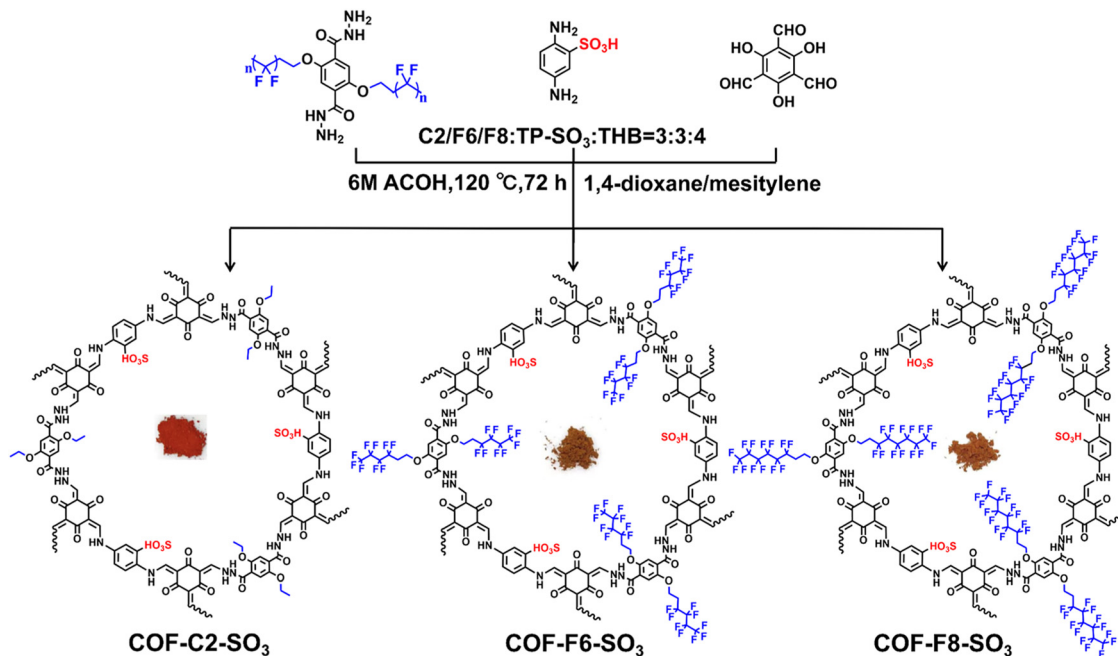


Fig. 29 Bottom-up strategy for the synthesis of COF-C2/Fx-SO₃ (x = 6, 8). Reproduced with permission from ref. 139 Copyright 2025 American Chemical Society.

acid was shown to function as a solid electrolyte with superprotonic conductivity of $1.06 \times 10^{-1} \text{ S cm}^{-1}$ at 423 K under anhydrous conditions. The combination of periodically aligned acidic sites and confined phosphoric acid enabled fast proton transport while suppressing acid volatilization, allowing stable operation within an electrochemical window of 2.4 V. Integration of this electrolyte into rechargeable proton batteries with vanadium hexacyanoferrate cathodes delivered a specific capacity of 101.8 mAh g^{-1} and retained 80% of the initial performance after 1000 cycles, underscoring the viability of COFs for solid-state proton batteries.²⁸⁸ Complementing this device-oriented approach, a parallel 2025 study demonstrated that nitrile- and ether-rich COFs can also act as active nanofillers in sulfonated poly(ether ether ketone), where their polar channels improve water management, mechanical integrity, and proton transport. The resulting composite membranes achieved conductivities up to 217 mS cm^{-1} at 80 °C and 90% RH and delivered a 2.3-fold increase in fuel-cell power density, highlighting the dual functionality of COFs as both structural and proton-conducting components.²⁸⁹ Structural innovations reported in 2025 further advanced COF-based proton conductors by directly addressing long-term durability and operation under harsh conditions. One representative strategy involved the development of perfluoroalkyl- and -SO₃H-functionalized COFs that deliberately mimic key architectural features of Nafion[®], combining hydrophobic stabilization with densely packed Brønsted acid sites. The introduction of fluorinated segments created water-resistant pore environments that suppressed acid leaching while markedly enhancing thermal and chemical stability, with frameworks remaining intact up to 350 °C and under strongly acidic conditions. Although moderate

conductivities were observed under humidified operation ($7.7 \times 10^{-3} \text{ S cm}^{-1}$ at 90 °C and 97% RH), phosphoric acid doping enabled efficient anhydrous proton transport, reaching $1.93 \times 10^{-2} \text{ S cm}^{-1}$ at 140 °C (Fig. 29).¹³⁹ Building on this durability-driven design, a complementary 2025 study introduced dual-acid-tailored COFs incorporating both sulfonic and phosphonic acid groups within a single framework. The cooperative interaction between these acidic functionalities generated dense hydrogen-bond networks that dramatically lowered activation energies (0.12–0.15 eV) and enabled ultrafast proton conduction up to 0.243 S cm^{-1} at 80 °C and 95% RH.²⁹¹ Together, these studies highlight how chemical synergy and pore-level engineering can reconcile high conductivity with long-term operational stability, paving the way toward practical next-generation proton exchange membranes.

The incorporation of crystalline and porous frameworks into sulfonated polyether ether ketone (SPEEK) has recently emerged as a promising strategy to overcome the conductivity-stability trade-off inherent to conventional polymer electrolytes. One representative example is the use of triazole-functionalized COFs capable of immobilizing phosphotungstic acid (HPW). Fan *et al.* demonstrated that blending HPW-loaded triazole COFs into SPEEK could simultaneously enhance proton conduction and stability. The triazole moieties provided anchoring sites that prevented HPW leaching while also facilitating hydrogen-bonded proton transport. The resulting hybrid membranes achieved conductivities above $10^{-2} \text{ S cm}^{-1}$ under humid conditions, clearly outperforming pristine SPEEK. This approach preserved the intrinsic acidity of HPW while mitigating its solubility, thereby combining the high conductivity of inorganic acids with the processability of polymer electrolytes.¹²⁸



In a different design, Yin and co-workers incorporated two-dimensional SCOF nanosheets into the SPEEK matrix. Unlike guest-loaded systems, here the $-\text{SO}_3\text{H}$ groups were covalently anchored to the COF network, establishing a robust and ordered network of Brønsted acid sites. The nanosheet morphology promoted homogeneous dispersion within the polymer phase, suppressed phase separation, and created continuous pathways for proton migration. These hybrid membranes achieved conductivity as high as $\sim 0.35 \text{ S cm}^{-1}$ at 80°C and full relative humidity, nearly twice that of pure SPEEK, while also displaying superior mechanical integrity and oxidative stability. This study highlighted the advantages of introducing nanoscale crystalline order and fixed sulfonic functionalities into polymer systems, offering a pathway to high-performance membranes with improved durability.²⁹²

More recently, Xu *et al.* reported the development of SPAFs blended into SPEEK. In contrast to crystalline COFs, SPAFs provide rigid but amorphous aromatic networks with high $-\text{SO}_3\text{H}$ density. Their incorporation restricted excessive swelling of the polymer while simultaneously enhancing water retention and proton conduction. The composite membranes maintained conductivity above $10^{-2} \text{ S cm}^{-1}$ over a wide humidity range and demonstrated improved dimensional and mechanical stability compared with pristine SPEEK. Furthermore, long-term durability tests indicated reduced performance degradation, underscoring the effectiveness of SPAFs in addressing the limitations of conventional sulfonated polymers (Fig. 30).²⁹³

Polybenzimidazole (PBI) has long been recognized as a robust high-temperature polymer for fuel cell membranes, particularly when doped with phosphoric acid, yet its relatively low proton conductivity and acid leaching have motivated the development of hybrid strategies. A recent approach involves the *in situ* growth of a sulfonated ionic COF (SCOF) directly within the PBI matrix, generating a dense distribution of $-\text{SO}_3\text{H}$ groups and ordered transport channels while forming acid-base ionic crosslinks with the imidazole sites of the polymer. This design effectively immobilizes phosphoric acid, enhances mechanical strength, and reduces proton transport barriers, with reported conductivities of $\sim 542 \text{ mS cm}^{-1}$ at 80°C under full humidity and stable performance in both electrochemical hydrogen compressor and single-cell tests.²⁹⁴ Another study integrated melamine-derived COFs into sulfonated PBI (SPBI)

to form nanocomposites where interfacial hydrogen bonding and higher ion exchange capacity (IEC) promoted continuous hydrophilic domains. At an optimized 15 wt% COF loading, the membranes exhibited conductivities around 0.108 S cm^{-1} at 80°C , improved tensile strength ($\sim 23 \text{ MPa}$), controlled water uptake, and delivered a peak power density of $\sim 394 \text{ mW cm}^{-2}$ at 80°C and full humidity, outperforming Nafion[®] 117 in cell tests, with oxidative stability retained at $\sim 91\text{--}93\%$.²⁹⁵ These results demonstrate that COF integration can significantly enhance the conductivity and durability of PBI-based membranes.

Recent efforts have also explored the integration of COFs with unconventional polymeric and biopolymer substrates, broadening the design space of hybrid proton-conducting membranes. One representative example is the combination of TpBd-2SO₃H COF nanosheets with sulfonated bacterial cellulose nanofibers (SBCNFs). The strong hydrogen bonding between COFs and cellulose fibers produced dense proton-conducting networks and significantly reinforced the mechanical strength of the resulting composites. The optimized SBCNF/TpBd-2SO₃H-0.5 membrane reached a conductivity of 145.7 mS cm^{-1} at 80°C and 100% RH, while maintaining tensile strength of nearly 88 MPa, much higher than pristine cellulose films. Importantly, the hybrid exhibited controlled swelling and improved dimensional stability, highlighting how biopolymer-COF systems can deliver both environmental compatibility and high performance for sustainable PEM development.²⁹⁶ A different design strategy involved the incorporation of ultrathin SCOF nanosheets into a Nafion[®] matrix. The two-dimensional morphology and dense $-\text{SO}_3\text{H}$ groups of SCOFs enabled homogeneous dispersion within Nafion[®], enhancing both phase separation and water retention. The optimum composite, containing 0.5 wt% SCOF loading, delivered a proton conductivity of 0.207 S cm^{-1} at 95°C and 100% RH, surpassing pristine Nafion[®]. Correspondingly, single-cell PEMFC tests exhibited a maximum power density of 1.024 W cm^{-2} , representing an $\sim 80\%$ improvement over unmodified Nafion[®]. The hybrid also maintained mechanical integrity and oxidative stability, demonstrating the ability of SCOF nanosheets to address Nafion's[®] long-standing weaknesses under harsh operating conditions.²⁹⁷

Most recently, a novel hybrid strategy was demonstrated by confining SCOFs within porous polytetrafluoroethylene (PTFE) substrates to produce highly flexible composite membranes.

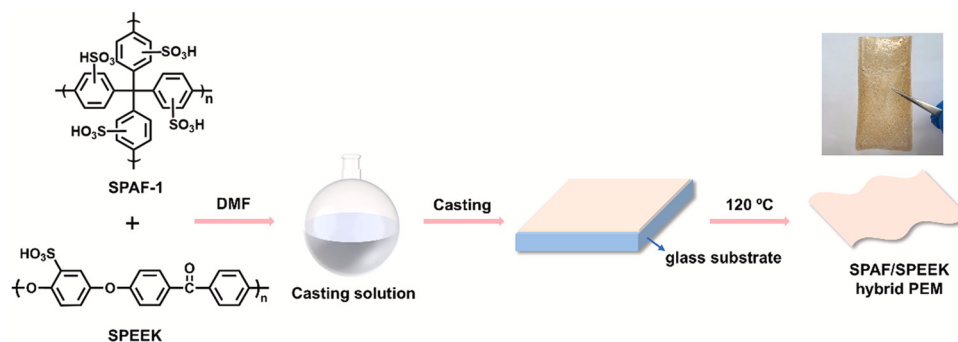


Fig. 30 Scheme for the preparation of SPAF/SPEEK hybrid membrane. Reproduced with permission from ref. 293 Copyright 2025 Elsevier.



Unlike conventional filler-polymer systems, the *in situ* growth of SCOFs in PTFE pores eliminated interfacial voids and ensured strong anchoring of the frameworks. The resulting membranes combined the hydrophobic stability of PTFE with the high acidity of SCOFs, achieving outstanding performance. The optimized SCOF/PTFE hybrid exhibited 0.249 S cm^{-1} conductivity at $80 \text{ }^\circ\text{C}$ and 95% RH and a peak fuel cell power density of 1195 mW cm^{-2} , exceeding the benchmark Nafion[®] 117 membrane. Mechanical testing confirmed tensile strengths above 30 MPa and excellent thermal stability up to $350 \text{ }^\circ\text{C}$, while durability studies showed minimal performance decay over extended operation. This dual-phase approach thus provides a scalable pathway to membranes with both exceptional proton conduction and robust long-term stability.²⁹⁸

5.2. Battery

COFs and POPs represent a molecularly precise platform for designing solid-state ionic conductors, directly addressing critical limitations in modern batteries. In lithium-ion batteries (LIBs), they enable safe, single-ion conduction by engineering anion-immobilized networks or nanoconfined ionic channels. Simultaneously, their ordered nanopores in vanadium redox flow batteries (VRFBs) solve the fundamental trade-off between high proton conductivity and low vanadium crossover. This section examines how strategic material design at the molecular level through fixed-anion architectures and tailored ionic nanoenvironments is advancing the performance of next-generation battery electrolytes and membranes. The promise of COF/POP-derived single-ion conductors for LIBs is illustrated by two complementary strategies that engineer Li^+ transport at the molecular level. An anionic PAF converted by chlorosulfonation and lithiation (Li-SPAF1) immobilizes the counter-anion on a rigid 3D diamond-type network, creating continuous hopping sites that raise the Li^+ transference number without sacrificing conductivity. Pressed Li-SPAF1 pellets deliver 0.102 mS cm^{-1} at $25 \text{ }^\circ\text{C}$ with $E_a \approx 0.21 \text{ eV}$, while DFT/NEB analyses attribute the performance to organized Li^+ migration along sulfonate-lined micropores (calculated barrier $\sim 0.33 \text{ eV}$), and Li||Li cells exhibit low interfacial resistance—evidence of favorable solid|Li contact.¹³⁴ A complementary route boosts ion dissociation and pathway accessibility by post-modifying a pyridine COF to a zwitterionic pyridinium-sulfobetaine that self-exfoliates into nanosheets. The resulting PS-CON hosts Li^+ in strongly solvating ionic environments and shortens diffusion lengths, yielding $2.19 \times 10^{-4} \text{ S cm}^{-1}$ at $25 \text{ }^\circ\text{C}$ (two orders higher than the parent COF), $t_{\text{Li}^+} = 0.46$, a widened electrochemical window to 4.05 V, and ultra-stable Li|Li cycling ($>1200 \text{ h}$, $<48 \text{ mV}$ overpotential), all consistent with robust interfacial kinetics and suppressed concentration polarization.²⁹⁹ Together, the fixed-anion 3D SPAF architecture and the zwitterionic, self-exfoliated 2D COF nanosheets delineate a clear design space for organic solid electrolytes: anchor anionic sites to direct Li^+ hopping, and engineer ionic backbones/nanosheet morphologies to enhance dissociation and lower transport barriers. These approaches are compatible with scalable processing while closing the conductivity gap to inorganic SSEs.

The challenge of simultaneously achieving high proton conductivity and low vanadium ion permeability in VRFBs has spurred the development of COF-based membrane designs. In 2022, sulfonated COF nanosheets were incorporated into a SPEEK matrix, where the highly ordered nanochannels facilitated efficient proton conduction while creating tortuous pathways that hindered VO^{2+} diffusion. The resulting hybrid membranes exhibited reduced crossover, improved coulombic efficiency, and stable long-term cycling compared to pristine polymer electrolytes.¹⁶⁴ A significant advance followed in 2023, when Angewandte Chemie reported the fabrication of self-standing SCOF membranes using macromolecular mediation. These thin crystalline films delivered well-defined proton-conducting channels with low activation energies, alongside high permselectivity in aqueous electrolytes, achieving energy efficiencies that surpassed those of conventional Nafion[®] membranes.³⁰⁰ In parallel, a study in the Journal of Applied Polymer Science developed hybrid membranes by immobilizing an imidazolium IL within TpPa-SO₃H COFs and dispersing the IL@COF filler into a sulfonated polymer matrix. This approach stabilized the IL, enhanced proton conductivity, suppressed vanadium ion crossover, and improved the mechanical properties of the membrane, resulting in higher energy efficiencies during VRFB operation.³⁰¹

COF-based membranes have been shown to significantly improve the trade-off between proton conductivity and vanadium ion selectivity in VRFBs. Incorporation of sulfonated frameworks into polymeric matrices generated hydrophilic domains enriched with Brønsted acid sites, which facilitated fast proton hopping while restricting VO^{2+} migration. The membranes displayed enhanced water retention and dimensional stability, leading to higher coulombic and energy efficiencies during prolonged cycling tests.¹⁰⁰ Structural engineering through hollow-fiber COFs further established directional nanochannels with intrinsic robustness, enabling energy efficiencies above 85% under elevated current densities and confirming the ability of ordered crystalline frameworks to sustain conductivity without compromising selectivity (Fig. 31).³⁰² A complementary approach involved blending SCOFs with ionomeric backbones to create homogeneous ionic domains, resulting in membranes that preserved high conductivity, suppressed vanadium diffusion, and maintained stable performance over extended charge-discharge cycles.²⁹⁰

6. Core challenges and key research directions

The impressive scope of applications for SPOPs outlined in this review underscores their significant potential. However, a critical assessment reveals that the field is still maturing from a discovery phase toward predictive design and practical implementation. Several interconnected core challenges must be systematically addressed to advance the field beyond proof-of-concept demonstrations.

(1) A predominant limitation is the frequent lack of molecular-level understanding. While high acid density is often



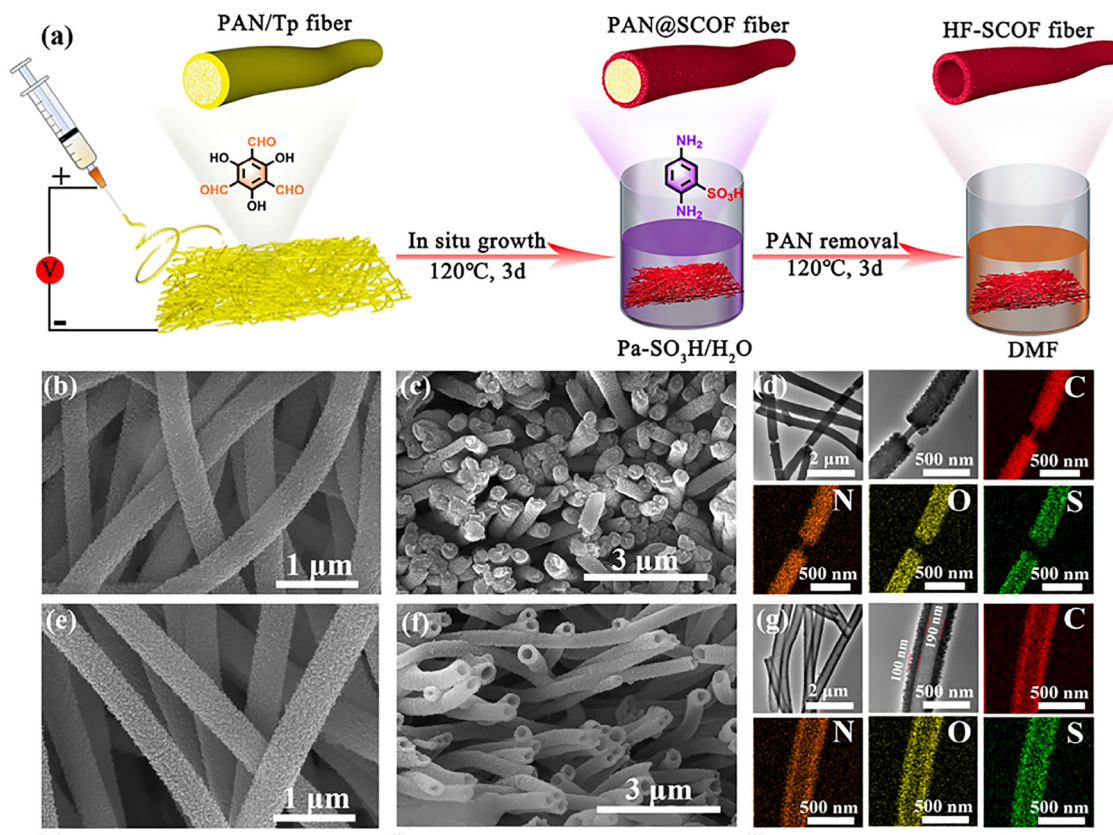


Fig. 31 Preparation, morphology, and structure of the hollow fiber HF-SCOF. a) Schematic preparation process; morphology of PAN@SCOF fiber of b) surface and c) cross-sectional SEM images, d) TEM and EDS images; morphology of HF-SCOF fiber of e) surface and f) cross-sectional SEM images, g) TEM and EDS images. Reproduced with permission from ref. 302 Copyright 2025 Wiley.

correlated with performance, the precise nature of the active site, its local chemical environment, effective acidity (pK_a distribution), spatial accessibility, and dynamic interaction with confined reactants remains poorly characterized. Mechanistic explanations often default to vague homogeneous analogies or generic references to “confinement effects”, without quantitative validation. Future work must prioritize the integration of advanced *in situ* and *operando* characterization techniques (e.g., solid-state NMR, X-ray absorption spectroscopy, vibrational spectroscopy) with computational modeling (DFT, molecular dynamics) to map acid-site strength, reactant diffusion barriers, and transition-state stabilization within engineered pores. Establishing predictive, quantitative models requires moving beyond reporting bulk properties (BET surface area, total acid density) to defining atomic-scale descriptors of active sites.

(2) The field suffers from a scarcity of direct, standardized comparisons between different SPOP classes (e.g., crystalline COF vs. amorphous HCP) or functionalization strategies (*de novo* vs. PSM) under identical application conditions. Consequently, fundamental questions about the advantages of long-range order, specific linkage chemistry, or pore topology for a given function remain unanswered. Optimal performance parameters (acid density vs. porosity preservation) are often found empirically. Systematic, side-by-side benchmarking studies are

urgently needed. Research should focus on controlled experiments that isolate key variables such as acid density in iso-structural frameworks or pore size with identical functionality to decouple their effects. This will establish true, causal structure–property relationships and enable rational material selection for targeted applications.

(3) For translation from laboratory to technology, long-term stability under real-world conditions is paramount but under-reported. The chemical stability of $-SO_3H$ groups and the hydrolytic/mechanical robustness of the framework in aqueous, thermal, or flowing reactant environments require rigorous assessment. Furthermore, the scalability of synthesis, especially for crystalline SPOPs, and their processability into application-ready forms (dense, defect-free membranes; structured monoliths; coated fabrics) present significant practical barriers. Research must adopt standardized stability testing protocols that mimic intended operational environments (e.g., accelerated stress tests for fuel cells). Parallel efforts should develop greener, scalable synthetic routes and innovative processing or composite strategies (e.g., mixed-matrix membranes, aerogels) to transform lab-scale powders into functional, engineered devices.

(4) Many studies showcase SPOPs in single-step reactions or the adsorption of model compounds. The next frontier requires moving from monofunctional materials to systems with deliberately



integrated, cooperative functionalities. The deliberate design of multivariate frameworks, where $-\text{SO}_3\text{H}$ sites are combined with orthogonal functional groups (*e.g.*, metal complexes, basic sites, redox-active moieties, or photosensitizers), is crucial. This will enable sophisticated applications such as acid–base or redox-acid tandem catalysis, stimuli-responsive capture/release, and dual ion–electron conduction. Achieving this requires sophisticated linker design, controlled multi-step PSM, and a deeper understanding of site–site interactions within the confined pore.

Addressing these core challenges will transform SPOP research from an exploratory, largely descriptive endeavor into a predictive, engineering-focused discipline. The following section outlines the specific opportunities and application horizons that become accessible once these foundational issues are resolved.

7. Outlook and future perspectives

The critical analysis above defines a clear roadmap for the field. By confronting the challenges of mechanistic understanding, benchmarking, stability, and complexity, researchers can unlock the next tier of opportunities for SPOPs. Their development has been propelled by the evolution of porous materials, from Nobel Prize-recognized MOFs to purely organic, covalently bonded frameworks. POPs, spanning amorphous HCPs to crystalline COFs, offer exceptional porosity, high surface areas, and outstanding stability. Yet, these intrinsic properties are merely the foundation. Strategic PSM, particularly sulfonation, tailors these materials to meet demanding application-specific needs, establishing a powerful structure–property–performance relationship where designed modifications yield synergistic effects and enhanced efficiency. This review has chronicled the transformative impact of creating SPOPs. By merging the robust, tunable porosity of POPs with the strong Brønsted acidity of $-\text{SO}_3\text{H}$ groups, SPOPs emerge as multifunctional platforms that surpass non-functionalized POPs and conventional solid acids like Amberlyst-15 or Nafion[®]. They are more than simple heterogeneous analogues of molecular acids like PTSA; their advanced functionality stems from the synergistic combination of nano-confinement, engineered porosity, and a high density of accessible acidic sites. A critical, unifying insight is the nonlinear role of acid loading, where a “more is better” approach often reduces performance due to pore blocking and hindered mass transport. Optimal performance results from a precise synergy of three key design elements: (i) High and accessible acidity: the density and strength of $-\text{SO}_3\text{H}$ sites available to reactants; (ii) Engineered nanoconfinement and porosity: pore architecture that governs mass transport, creates confined nanoreactor environments, and imparts selectivity; and (iii) Tailored surface chemistry and multifunctionality: the ability to incorporate co-functional groups (*e.g.*, $-\text{NH}_2$, metal complexes) to enable cooperative mechanisms or cascade processes. This design principle empowers SPOPs across three critical technological domains: heterogeneous catalysis, where they bridge homogeneous acid potency with heterogeneous practicality for biodiesel production, biomass conversion, and

CO_2 utilization; advanced environmental remediation and separation, where they act as highly efficient adsorbents for pollutants and selective captors for gases and critical resources; and next-generation electrochemical energy technologies, where they decouple ion conductivity from stability in fuel cells, redox flow batteries, and solid-state batteries. To realize this potential, future research should focus on several key avenues that directly address the core challenges:

- Multifunctional system design: integrating $-\text{SO}_3\text{H}$ sites with metal complexes, basic sites, or photoresponsive units within a single framework for tandem catalysis and smart adsorbents.
- Stability and processability: rigorously testing long-term hydrolytic and mechanical stability under harsh conditions and developing scalable processing routes to create membranes, monoliths, and coated fabrics.
- Application in emerging energy technologies: leveraging innate proton conductivity to replace fluorinated membranes (*e.g.*, Nafion[®]) in fuel cells and flow batteries, and exploring roles as solid electrolytes.
- Advanced environmental remediation: deploying SPOPs for the selective removal of emerging contaminants like pharmaceuticals and PFAS from complex water streams.
- Defect engineering: introducing controlled defects to modulate porosity, enhance active-site accessibility, and create new functionalities.
- Complex and multivariate architectures: constructing hierarchical heterostructures (*e.g.*, POP-on-POP, POP-on-MOF) to unlock synergistic effects in catalysis, separation, sensing, and energy storage.

In conclusion, SPOPs is becoming an emerging class of functional porous materials with considerable potential in many areas. They provide a flexible and powerful platform that goes beyond the traditional concept of solid acid. By improving synthetic control, deepening fundamental understanding, and actively exploring new application areas, SPOPs are ready to move from laboratory to commercial products becoming essential materials that support solutions for sustainable chemistry, environmental protection, and energy technology. Their future is not just promising; it is necessary given the urgency of developing renewable energy technologies and replacing fluorinated polymers.

Conflicts of interest

There are no conflicts to declare.

Data availability

The present review has not generated new scientific data. Any issue regarding the present submission will be addressed by the corresponding authors.

Acknowledgements

This work was supported by MICINN (PID2024-15520NB-I00, PID2022-138470NB-I00, PID2021-122299NB-I00, TED2021-129999B-



C32, TED2021-130470B-100), VIRMAT projects in response to COVID-19 financed by the ERDF-REACT-EU resources, MSCD-Doctoral Network 2023 – “Photocatalytic Industrial Applications” (PIA) (Reference: 101168878), “Comunidad de Madrid” for European Structural Funds (S2018/NMT-4367), Laboratorios inteligentes para la ciencia del futuro: descubrimiento de materiales avanzados para fotosíntesis artificial (FOTOART5.0-CM) (Reference: TEC2024/TEC-308) and proyectos sinérgicos I + D (Y2020/NMT-6469). The authors also gratefully acknowledge the support of Lorestan University (LU).

References

- 1 T. Zhang, G. Xing, W. Chen and L. Chen, *Mater. Chem. Front.*, 2020, **4**, 332–353.
- 2 P. Kaur, J. T. Hupp and S. T. Nguyen, *ACS Catal.*, 2011, **1**, 819–835.
- 3 A. P. Côté, A. I. Benin, N. W. Ockwig, M. O’Keeffe, A. J. Matzger and O. M. Yaghi, *Science*, 2005, **310**, 1166–1170.
- 4 T. Zhang, V. G. Gregoriou, N. Gasparini and C. L. Chochos, *Chem. Soc. Rev.*, 2022, **51**, 4465–4483.
- 5 A. K. Sahoo, D. Sarkar, A. Kumar and J. N. Moorthy, *Chem. – Eur. J.*, 2025, **31**, e01267.
- 6 S. Kitagawa, R. Kitaura and S.-I. Noro, *Angew. Chem., Int. Ed.*, 2004, **43**, 2334–2375.
- 7 M. Kondo, T. Yoshitomi, H. Matsuzaka, S. Kitagawa and K. Seki, *Angew. Chem., Int. Ed. Engl.*, 1997, **36**, 1725–1727.
- 8 B. F. Hoskins and R. Robson, *J. Am. Chem. Soc.*, 1989, **111**, 5962–5964.
- 9 O. M. Yaghi, G. Li and H. Li, *Nature*, 1995, **378**, 703–706.
- 10 X. Feng, X. Ding and D. Jiang, *Chem. Soc. Rev.*, 2012, **41**, 6010–6022.
- 11 A. R. Oveisi, K. Zhang, A. Khorramabadi-zad, O. K. Farha and J. T. Hupp, *Sci. Rep.*, 2015, **5**, 10621.
- 12 J. Huang and S. R. Turner, *Polym. Rev.*, 2018, **58**, 1–41.
- 13 V. Davankov, S. Rogozhin and M. Tsyurupa, *Mezhdunar. Kniga*, 1973, **15**, 463–465.
- 14 V. A. Davankov and M. P. Tsyurupa, *React. Polym.*, 1990, **13**, 27–42.
- 15 M. P. Tsyurupa, Z. K. Blinnikova, N. A. Proskurina, A. V. Pastukhov, L. A. Pavlova and V. A. Davankov, *Nanotechnol. Russ.*, 2009, **4**, 665–675.
- 16 N. B. McKeown, S. Makhseed and P. M. Budd, *Chem. Commun.*, 2002, 2780–2781.
- 17 P. M. Budd, B. S. Ghanem, S. Makhseed, N. B. McKeown, K. J. Msayib and C. E. Tattershall, *Chem. Commun.*, 2004, 230–231.
- 18 N. B. McKeown and P. M. Budd, *Chem. Soc. Rev.*, 2006, **35**, 675–683.
- 19 N. B. McKeown, *Polymer*, 2020, **202**, 122736.
- 20 A. López-Magano, S. Daliran, A. R. Oveisi, R. Mas-Ballesté, A. Dhakshinamoorthy, J. Alemán, H. Garcia and R. Luque, *Adv. Mater.*, 2023, **35**, 2209475.
- 21 P. J. Waller, F. Gándara and O. M. Yaghi, *Acc. Chem. Res.*, 2015, **48**, 3053–3063.
- 22 S.-Y. Ding and W. Wang, *Chem. Soc. Rev.*, 2013, **42**, 548–568.
- 23 K. T. Tan, S. Ghosh, Z. Wang, F. Wen, D. Rodríguez-San-Miguel, J. Feng, N. Huang, W. Wang, F. Zamora, X. Feng, A. Thomas and D. Jiang, *Nat. Rev. Methods Primers*, 2023, **3**, 1.
- 24 R. P. Bisbey and W. R. Dichtel, *ACS Cent. Sci.*, 2017, **3**, 533–543.
- 25 B. Liu, Y. Wu, L. Wang, H.-L. Jiang and Q. Li, *Nat. Chem.*, 2026, **18**, 64–172.
- 26 J.-X. Jiang, F. Su, A. Trewin, C. D. Wood, N. L. Campbell, H. Niu, C. Dickinson, A. Y. Ganin, M. J. Rosseinsky, Y. Z. Khimiyak and A. I. Cooper, *Angew. Chem., Int. Ed.*, 2007, **46**, 8574–8578.
- 27 Y. Xu, S. Jin, H. Xu, A. Nagai and D. Jiang, *Chem. Soc. Rev.*, 2013, **42**, 8012–8031.
- 28 J.-S. M. Lee and A. I. Cooper, *Chem. Rev.*, 2020, **120**, 2171–2214.
- 29 J. Kim, C. M. Moisanu, C. N. Gannett, A. Halder, J. J. Fuentes-Rivera, S. H. Majer, K. M. Lancaster, A. C. Forse, H. D. Abruña and P. J. Milner, *Chem. Mater.*, 2021, **33**, 8334–8342.
- 30 L. Tan and B. Tan, *Chem. Soc. Rev.*, 2017, **46**, 3322–3356.
- 31 P. Kuhn, M. Antonietti and A. Thomas, *Angew. Chem., Int. Ed.*, 2008, **47**, 3450–3453.
- 32 C. Krishnaraj, H. S. Jena, K. Leus and P. Van Der Voort, *Green Chem.*, 2020, **22**, 1038–1071.
- 33 L. Liao, M. Li, Y. Yin, J. Chen, Q. Zhong, R. Du, S. Liu, Y. He, W. Fu and F. Zeng, *ACS Omega*, 2023, **8**, 4527–4542.
- 34 B. Darsanj, E. Heydari-Bafrooei and M. Dinari, *Energy Fuels*, 2025, **39**, 4450–4457.
- 35 S.-Y. Yu, J. Mahmood, H.-J. Noh, J.-M. Seo, S.-M. Jung, S.-H. Shin, Y.-K. Im, I.-Y. Jeon and J.-B. Baek, *Angew. Chem., Int. Ed.*, 2018, **57**, 8438–8442.
- 36 M. Liu, L. Guo, S. Jin and B. Tan, *J. Mater. Chem. A*, 2019, **7**, 5153–5172.
- 37 J. Tan, S. Namuangruk, W. Kong, N. Kungwan, J. Guo and C. Wang, *Angew. Chem., Int. Ed.*, 2016, **55**, 13979–13984.
- 38 B. Zhang, M. Wei, H. Mao, X. Pei, S. A. Alshimri, J. A. Reimer and O. M. Yaghi, *J. Am. Chem. Soc.*, 2018, **140**, 12715–12719.
- 39 Y. Zhai, G. Liu, F. Jin, Y. Zhang, X. Gong, Z. Miao, J. Li, M. Zhang, Y. Cui, L. Zhang, Y. Liu, H. Zhang, Y. Zhao and Y. Zeng, *Angew. Chem., Int. Ed.*, 2019, **58**, 17679–17683.
- 40 T. Ben, H. Ren, S. Ma, D. Cao, J. Lan, X. Jing, W. Wang, J. Xu, F. Deng, J. M. Simmons, S. Qiu and G. Zhu, *Angew. Chem., Int. Ed.*, 2009, **48**, 9457–9460.
- 41 T. Ben and S. Qiu, *CrystEngComm*, 2013, **15**, 17–26.
- 42 Y. Tian and G. Zhu, *Chem. Rev.*, 2020, **120**, 8934–8986.
- 43 E. Karakhanov, Y. Kardasheva, L. Kulikov, A. Maximov, A. Zolotukhina, M. Vinnikova and A. Ivanov, *Catalysts*, 2016, **6**, 122.
- 44 Y. Yuan, F. Sun, H. Ren, X. Jing, W. Wang, H. Ma, H. Zhao and G. Zhu, *J. Mater. Chem.*, 2011, **21**, 13498–13502.
- 45 J. Germain, J. M. J. Fréchet and F. Svec, *J. Mater. Chem.*, 2007, **17**, 4989–4997.
- 46 H. Xu, J. Gao and D. Jiang, *Nat. Chem.*, 2015, **7**, 905–912.
- 47 N. Naz, M. H. Manzoor, S. M. G. Naqvi, U. Ehsan, M. Aslam and F. Verpoort, *Appl. Mater. Today*, 2024, **38**, 102198.



- 48 S. Zhang, Q. Yang, C. Wang, X. Luo, J. Kim, Z. Wang and Y. Yamauchi, *Adv. Sci.*, 2018, **5**, 1801116.
- 49 J. Perego, S. Bracco, A. Comotti, D. Piga, I. Bassanetti and P. Sozzani, *Angew. Chem., Int. Ed.*, 2021, **60**, 6117–6123.
- 50 Z. Xiang, X. Zhou, C. Zhou, S. Zhong, X. He, C. Qin and D. Cao, *J. Mater. Chem.*, 2012, **22**, 22663–22669.
- 51 N. A. Babujohn, A. Eluri and V. P. Nabeela, *Chem. Eng. J.*, 2023, **464**, 142459.
- 52 L. Yuan, N. Zheng, T. Yang, A. Li, Y. Yuan, J. Hua, L. Li and C. Zhou, *J. Taiwan Inst. Chem. Eng.*, 2023, **144**, 104733.
- 53 W. Lu, D. Yuan, J. Sculley, D. Zhao, R. Krishna and H.-C. Zhou, *J. Am. Chem. Soc.*, 2011, **133**, 18126–18129.
- 54 W. Lu, D. Yuan, D. Zhao, C. I. Schilling, O. Plietzsch, T. Muller, S. Bräse, J. Guenther, J. Blümel, R. Krishna, Z. Li and H.-C. Zhou, *Chem. Mater.*, 2010, **22**, 5964–5972.
- 55 X. Yang, L. Zou and H.-C. Zhou, *Polymer*, 2017, **126**, 303–307.
- 56 J. Y. Jang, T. M. D. Le, J. H. Ko, Y.-J. Ko, S. M. Lee, H. J. Kim, J. H. Jeong, T. Thambi, D. S. Lee and S. U. Son, *Chem. Mater.*, 2019, **31**, 300–304.
- 57 H. J. Jeon, J. H. Choi, Y. Lee, K. M. Choi, J. H. Park and J. K. Kang, *Adv. Energy Mater.*, 2012, **2**, 225–228.
- 58 H. S. Choi, H. J. Jeon, J. H. Choi, G.-H. Lee and J. K. Kang, *Nanoscale*, 2015, **7**, 18923–18927.
- 59 A. O. Mousa, M. G. Mohamed, C.-H. Chuang and S.-W. Kuo, *Polymers*, 2023, **15**, 1891.
- 60 B. Yu, Y. Tao, X. Yao, Y. Jin, S. Liu, T. Xu, H. Wang, H. Wu, W. Zhou, X. Zhou, X. Ding, X. Wang, X. Xiao, Y.-B. Zhang and J. Jiang, *J. Am. Chem. Soc.*, 2024, **146**, 28932–28940.
- 61 L. Zhu, Y. Cao, T. Xu, H. Yang, L. Wang, L. Dai, F. Pan, C. Chen and C. Si, *Energy Environ. Sci.*, 2025, **18**, 5675–5739.
- 62 K. S. Song, P. W. Fritz and A. Coskun, *Chem. Soc. Rev.*, 2022, **51**, 9831–9852.
- 63 S. Fajal, S. Dutta and S. K. Ghosh, *Mater. Horiz.*, 2023, **10**, 4083–4138.
- 64 B. Lopez-Iglesias, F. Suárez-García, C. Aguilar-Lugo, A. González Ortega, C. Bartolomé, J. M. Martínez-Ilarduya, J. G. de la Campa, Á. E. Lozano and C. Álvarez, *ACS Appl. Mater. Interfaces*, 2018, **10**, 26195–26205.
- 65 J.-X. Hu, H. Shang, J.-G. Wang, L. Luo, Q. Xiao, Y.-J. Zhong and W.-D. Zhu, *Ind. Eng. Chem. Res.*, 2014, **53**, 11828–11837.
- 66 M. Gomez-Suarez, Y. Chen and J. Zhang, *Polym. Chem.*, 2023, **14**, 4000–4032.
- 67 Q. Li, Y. Wu, X. Tan, B. Ruan, Y. Yu, X. Chen, B. Zhao and L. Huang, *ACS Appl. Polym. Mater.*, 2025, **7**, 15429–15442.
- 68 M. C. Nilavu, H. Aggarwal and N. Rajesh, *ACS Appl. Eng. Mater.*, 2025, **3**, 2007–2019.
- 69 A. Hassan and N. Das, *ACS Appl. Eng. Mater.*, 2024, **2**, 56–66.
- 70 H. Wang, Y. Zhai, Y. Li, Y. Cao, B. Shi, R. Li, Z. Zhu, H. Jiang, Z. Guo, M. Wang, L. Chen, Y. Liu, K.-G. Zhou, F. Pan and Z. Jiang, *Nat. Commun.*, 2022, **13**, 7123.
- 71 T. Ashirov, J. Lim, A. Robles, T. Puangsamlee, P. W. Fritz, A. Crochet, X. Wang, C. Hewson, P. Iacomì, O. Š. Miljanić and A. Coskun, *Angew. Chem., Int. Ed.*, 2025, **64**, e202423809.
- 72 M. Debruyne, V. Van Speybroeck, P. Van Der Voort and C. V. Stevens, *Green Chem.*, 2021, **23**, 7361–7434.
- 73 W. Qu, J. Hu, H. Xu, C. Chen, Q. Sun, J. Chen and X. Jia, *Catal. Sci. Technol.*, 2025, **15**, 7004–7008.
- 74 M. C. Borrallo-Aniceto, M. Pintado-Sierra, A. Valverde-González, U. Díaz, F. Sánchez, E. M. Maya and M. Iglesias, *Green Chem.*, 2024, **26**, 1975–1983.
- 75 M. Debruyne, N. Raeymackers, H. Vrielinck, S. Radhakrishnan, E. Breynaert, M. Delaey, A. Laemont, K. Leus, J. Everaert, H. Rijckaert, D. Poelman, R. Morent, N. De Geyter, P. Van Der Voort, V. Van Speybroeck, C. V. Stevens and T. S. A. Heugebaert, *ChemCatChem*, 2024, **16**, e202301205.
- 76 A. Karn, F. Sama, V. V. Chernyshev and J. N. Moorthy, *ACS Appl. Polym. Mater.*, 2025, **7**, 9120–9130.
- 77 H. Li, F. Pan, C. Qin, T. Wang and K.-J. Chen, *Adv. Energy Mater.*, 2023, **13**, 2301378.
- 78 R. K. Totten, Y.-S. Kim, M. H. Weston, O. K. Farha, J. T. Hupp and S. T. Nguyen, *J. Am. Chem. Soc.*, 2013, **135**, 11720–11723.
- 79 J. Mondal, Q. T. Trinh, A. Jana, W. K. H. Ng, P. Borah, H. Hirao and Y. Zhao, *ACS Appl. Mater. Interfaces*, 2016, **8**, 15307–15319.
- 80 J. H. Ko, N. Kang, N. Park, H.-W. Shin, S. Kang, S. M. Lee, H. J. Kim, T. K. Ahn and S. U. Son, *ACS Macro Lett.*, 2015, **4**, 669–672.
- 81 C. Sarkar, C. Biswas, R. Paul and J. Mondal, *Chem. Commun.*, 2025, **61**, 12283–12300.
- 82 M. A. Shirani, M. Afshari, I. Romero-Muñiz, A. E. Platero-Prats, M. Dinari and F. Zamora, *J. Mater. Chem. A*, 2025, **13**, 19172–19192.
- 83 J. Guo and D. Jiang, *ACS Cent. Sci.*, 2020, **6**, 869–879.
- 84 Q. Yu, Y.-Z. Cheng, Z. Li, D.-H. Yang, Q. Bo Meng and B.-H. Han, *Chem. Eng. J.*, 2022, **442**, 136275.
- 85 A. R. Oveisi, S. Heydarani, S. Daliran, E. Sanchooli, M. Khajeh, J. Albero, C. Molinet-Chinaglia and H. Garcia, *Adv. Funct. Mater.*, 2025, **35**, e09095.
- 86 Y. Zhu and W. Zhang, *Chem. Sci.*, 2014, **5**, 4957–4961.
- 87 A. Giri, Y. Khakre, G. Shreeraj, T. K. Dutta, S. Kundu and A. Patra, *J. Mater. Chem. A*, 2022, **10**, 17077–17121.
- 88 S. Oudi, A. R. Oveisi, S. Daliran, M. Khajeh, A. Dhakshinamoorthy and H. Garcia, *Nanomaterials*, 2023, **13**, 558.
- 89 H. Bildirir, V. G. Gregoriou, A. Avgeropoulos, U. Scherf and C. L. Chochos, *Mater. Horiz.*, 2017, **4**, 546–556.
- 90 Y. Feng, H. Zhao, J. Liu, Z. Song, W. Che, K. Ma and Y. Wang, *Constr. Build. Mater.*, 2023, **409**, 134042.
- 91 M. G. Mohamed, A. F. M. El-Mahdy, M. G. Kotp and S.-W. Kuo, *Mater. Adv.*, 2022, **3**, 707–733.
- 92 P. Zhao, L. Tian, Y. Guo, B. Lv, X. Mao, T. Li, J. Cui, J. Guo and B. Yang, *Eur. Polym. J.*, 2022, **162**, 110918.
- 93 Q. Sun, B. Aguila, Y. Song and S. Ma, *Acc. Chem. Res.*, 2020, **53**, 812–821.
- 94 A. Facchetti, *Chem. Mater.*, 2011, **23**, 733–758.
- 95 W. Liu, Y. Yang, L. Guo, J. Di, C. Hon Lau, M. V. Bermeshev and L. Shao, *Chem. Eng. J.*, 2024, **498**, 155569.
- 96 A. Modak, A. Ghosh, A. R. Mankar, A. Pandey, M. Selvaraj, K. K. Pant, B. Chowdhury and A. Bhaumik, *ACS Sustainable Chem. Eng.*, 2021, **9**, 12431–12460.
- 97 J. Ozdemir, I. Mosleh, M. Abolhassani, L. F. Greenlee, R. R. Beitle and M. H. Beyzavi, *Front. Energy Res.*, 2019, **7**, 77.



- 98 Y. Zhu, P. Xu, X. Zhang and D. Wu, *Chem. Soc. Rev.*, 2022, **51**, 1377–1414.
- 99 X. H. Xiong, Y. Tao, Z. W. Yu, L. X. Yang, L. J. Sun, Y. L. Fan and F. Luo, *Chem. Eng. J.*, 2020, **384**, 123240.
- 100 B. Pang, R. Du, W. Chen, F. Cui, N. Wang, H. Zhao, G. Xie, L. Tiantian, G. He and X. Wu, *Energy Storage Mater.*, 2024, **67**, 103293.
- 101 Y. Huang, L. Zhi, C. Cheng, Q. Wei, S. Sun and C. Zhao, *Colloids Surf., B*, 2025, **253**, 114716.
- 102 T.-X. Wang, H.-P. Liang, D. A. Anito, X. Ding and B.-H. Han, *J. Mater. Chem. A*, 2020, **8**, 7003–7034.
- 103 X. Liu, C.-F. Liu, S. Xu, T. Cheng, S. Wang, W.-Y. Lai and W. Huang, *Chem. Soc. Rev.*, 2022, **51**, 3181–3225.
- 104 N. Singh, S. Son, J. An, I. Kim, M. Choi, N. Kong, W. Tao and J. S. Kim, *Chem. Soc. Rev.*, 2021, **50**, 12883–12896.
- 105 D. Luo, M. Li, Q. Ma, G. Wen, H. Dou, B. Ren, Y. Liu, X. Wang, L. Shui and Z. Chen, *Chem. Soc. Rev.*, 2022, **51**, 2917–2938.
- 106 S. Wang, H. Li, H. Huang, X. Cao, X. Chen and D. Cao, *Chem. Soc. Rev.*, 2022, **51**, 2031–2080.
- 107 K. Miller, J. M. Gayle, S. Roy, M. H. Abdellah, R. Hardian, L. Cseri, P. G. Demingos, H. R. Nadella, F. Lee, M. Tripathi, S. Gupta, G. Guo, S. Bhattacharyya, X. Wang, A. B. Dalton, A. Garg, C. V. Singh, R. Vajtai, G. Szekely and P. Ajayan, *Small*, 2024, **20**, 2401269.
- 108 M. G. Rabbani, R. K. Sasse, S. Behera, P. Jena, J. Liu, P. K. Thallapally, T. Islamoglu, M. K. Shehab, M. M. Kaid, O. K. Farha and H. M. El-Kaderi, *Langmuir*, 2024, **40**, 8024–8034.
- 109 H. Li, Z. Zhou, T. Ma, K. Wang, H. Zhang, A. H. Alawadhi and O. M. Yaghi, *J. Am. Chem. Soc.*, 2024, **146**, 35486–35492.
- 110 F. Gandomkar, M. Khajeh, A. R. Oveisi, P. Karimi and H. Garcia, *Polymer*, 2025, **316**, 127858.
- 111 A. H. Zare, M. Khajeh, A. R. Oveisi, S. Daliran and M. Ghaffari-Moghaddam, *J. Polym. Environ.*, 2024, **32**, 5771–5782.
- 112 D. Karimi, M. Khajeh, A. R. Oveisi, M. Bohlooli, A. Khatibi, R. S. Neyband and R. Luque, *Environ. Pollut.*, 2023, **334**, 122109.
- 113 N. Nouruzi, M. Dinari, B. Gholipour, M. Afshari and S. Rostamnia, *ACS Appl. Nano Mater.*, 2022, **5**, 6241–6248.
- 114 H. Ali, Y. Orooji, A. Y. A. Alzahrani, M. H. Al Mughram, A. M. Abu-Dief, I. Omar, A. Hayat, D. Yue and Y. Xu, *Coord. Chem. Rev.*, 2025, **543**, 216884.
- 115 J. H. Kim, D. W. Kang, H. Yun, M. Kang, N. Singh, J. S. Kim and C. S. Hong, *Chem. Soc. Rev.*, 2022, **51**, 43–56.
- 116 A. Munyentwali, C. Li, H. Li and Q. Yang, *Chem. – Asian J.*, 2021, **16**, 2041–2047.
- 117 M. B. Karimi, F. Mohammadi and K. Hooshyari, *Int. J. Hydrogen Energy*, 2019, **44**, 28919–28938.
- 118 F. Liu, X. Meng, Y. Zhang, L. Ren, F. Nawaz and F.-S. Xiao, *J. Catal.*, 2010, **271**, 52–58.
- 119 S. Bhunia, B. Banerjee and A. Bhaumik, *Chem. Commun.*, 2015, **51**, 5020–5023.
- 120 Y. Peng, Z. Hu, Y. Gao, D. Yuan, Z. Kang, Y. Qian, N. Yan and D. Zhao, *ChemSusChem*, 2015, **8**, 3208–3212.
- 121 N. Park, Y. N. Lim, S. Y. Kang, S. M. Lee, H. J. Kim, Y.-J. Ko, B. Y. Lee, H.-Y. Jang and S. U. Son, *ACS Macro Lett.*, 2016, **5**, 1322–1326.
- 122 M. G. Goesten, A. Szécsényi, M. F. de Lange, A. V. Bavykina, K. B. S. S. Gupta, F. Kapteijn and J. Gascon, *ChemCatChem*, 2016, **8**, 961–967.
- 123 R. Gomes, P. Bhanja and A. Bhaumik, *J. Mol. Catal. A: Chem.*, 2016, **411**, 110–116.
- 124 Y. Peng, G. Xu, Z. Hu, Y. Cheng, C. Chi, D. Yuan, H. Cheng and D. Zhao, *ACS Appl. Mater. Interfaces*, 2016, **8**, 18505–18512.
- 125 P. Bhanja, A. Palui, S. Chatterjee, Y. V. Kaneti, J. Na, Y. Sugahara, A. Bhaumik and Y. Yamauchi, *ACS Sustainable Chem. Eng.*, 2020, **8**, 2423–2432.
- 126 S.-J. Yang, X. Ding and B.-H. Han, *Langmuir*, 2018, **34**, 7640–7646.
- 127 Z. Li, Z. Liu, H. Li, M. Hasan, A. Suwansontorn, G. Du, D. Wang, Y. Zhang and Y. Nagao, *ACS Appl. Polym. Mater.*, 2020, **2**, 3267–3273.
- 128 C. Fan, H. Wu, Y. Li, B. Shi, X. He, M. Qiu, X. Mao and Z. Jiang, *Solid State Ion.*, 2020, **349**, 115316.
- 129 Y. Lei, M. Zhang, G. Leng, C. Ding and Y. Ni, *Microporous Mesoporous Mater.*, 2020, **299**, 110110.
- 130 W. Zhao, Y. Jiao, R. Gao, L. Wu, S. Cheng, Q. Zhuang, A. Xie and W. Dong, *Chem. Eng. J.*, 2020, **391**, 123591.
- 131 M. A. Kalinina, L. A. Kulikov, K. A. Cherednichenko, A. L. Maximov and E. A. Karakhanov, *Pet. Chem.*, 2021, **61**, 1061–1070.
- 132 B.-J. Yao, W.-X. Wu, L.-G. Ding and Y.-B. Dong, *J. Org. Chem.*, 2021, **86**, 3024–3032.
- 133 Y. Zhang, C. Li, Z. Liu, Y. Yao, M. M. Hasan, Q. Liu, J. Wan, Z. Li, H. Li and Y. Nagao, *CrystEngComm*, 2021, **23**, 6234–6238.
- 134 S. Lee, J. Jeong, J. Moon, M. Kim, K. V. L. Amarasinghe, K. Yoon Chung, H.-D. Lim and D. Whang, *Chem. Eng. J.*, 2021, **424**, 130527.
- 135 H. Dong, Z. Zhu, K. Li, Q. Li, W. Ji, B. He, J. Li and X. Ma, *J. Membr. Sci.*, 2021, **635**, 119440.
- 136 H. Zhou, C. Rayer, A. R. Antonangelo, N. Hawkins and M. Carta, *ACS Appl. Mater. Interfaces*, 2022, **14**, 20997–21006.
- 137 L. Zhai, Y. Yao, B. Ma, M. M. Hasan, Y. Han, L. Mi, Y. Nagao and Z. Li, *Macromol. Rapid Commun.*, 2022, **43**, 2100590.
- 138 Z. Shao, X. Xue, K. Gao, J. Chen, L. Zhai, T. Wen, S. Xiong, H. Hou and L. Mi, *J. Mater. Chem. A*, 2023, **11**, 3446–3453.
- 139 Y. Du, Y. Yang, Z. Shan, X. Xu, H. Xu, B. He, Y. Liu, X. Zhang, B. Xu and G. Zhang, *ACS Appl. Nano Mater.*, 2025, **8**, 6747–6754.
- 140 P. S. Tully, in Kirk-Othmer Encyclopedia of Chemical Technology, *Sulfonic Acids*, Wiley, New York, 2000, DOI: [10.1002/0471238961.1921120620211212.a01](https://doi.org/10.1002/0471238961.1921120620211212.a01).
- 141 B. Bitá, *Curr. Org. Chem.*, 2011, **15**, 3091–3097.
- 142 S. Pal, D. Das and S. Bhunia, *Org. Biomol. Chem.*, 2024, **22**, 1527–1579.
- 143 J. C. Wu, B. H. Wang, D. L. Zhang, G. F. Song, J. T. Yuan and B. F. Liu, *J. Chem. Technol. Biotechnol.*, 2001, **76**, 619–623.
- 144 M. P. D. Mahindaratne and K. Wimalasena, *J. Org. Chem.*, 1998, **63**, 2858–2866.
- 145 S. Kang, G. Zhang, X. Yang, H. Yin, X. Fu, J. Liao, J. Tu, X. Huang, F. G. F. Qin and Y. Xu, *Energy Fuels*, 2017, **31**, 2847–2854.



- 146 J. P. Guthrie, *Can. J. Chem.*, 1978, **56**, 2342–2354.
- 147 L. Jin, Z. Ren, C. Yang, J. Li, Y. Shan and P. Liu, *Appl. Catal., A*, 2025, **698**, 120236.
- 148 A. Corma, *Curr. Opin. Solid State Mater. Sci.*, 1997, **2**, 63–75.
- 149 G. De Crescenzo, B. de Santos, M. J. Capitán, J. Álvarez, S. Cabrera, A. Fraile, M. Blanco and J. Alemán, *J. Mater. Chem. A*, 2025, **13**, 23928–23938.
- 150 C.-J. Wu, X.-Y. Li, T.-R. Li, M.-Z. Shao, L.-J. Niu, X.-F. Lu, J.-L. Kan, Y. Geng and Y.-B. Dong, *J. Am. Chem. Soc.*, 2022, **144**, 18750–18755.
- 151 C.-J. Wu, M.-Z. Shao, J.-L. Kan, W.-J. Liang, T.-R. Li, L.-J. Niu, H.-J. Wang, Y. Geng and Y.-B. Dong, *ACS Mater. Lett.*, 2024, **6**, 5016–5022.
- 152 G.-B. Wang, Y.-J. Wang, J.-L. Kan, K.-H. Xie, H.-P. Xu, F. Zhao, M.-C. Wang, Y. Geng and Y.-B. Dong, *J. Am. Chem. Soc.*, 2023, **145**, 4951–4956.
- 153 S. Daliran, M. Blanco, A. Dhakshinamoorthy, A. R. Oveisi, J. Alemán and H. García, *Adv. Funct. Mater.*, 2024, **34**, 2312912.
- 154 A. Blocher, F. Mayer, P. Schweng, T. M. Tikovits, N. Yousefi and R. T. Woodward, *Mater. Adv.*, 2022, **3**, 6335–6342.
- 155 S. Ge, K. Wei, W. Peng, R. Huang, E. Akinlabi, H. Xia, M. W. Shahzad, X. Zhang, B. B. Xu and J. Jiang, *Chem. Soc. Rev.*, 2024, **53**, 11259–11302.
- 156 R. Li, X. Tang, J. Wu, K. Zhang, Q. Zhang, J. Wang, J. Zheng, S. Zheng, J. Fan, W. Zhang, X. Li and S. Cai, *Chem. Eng. J.*, 2023, **464**, 142706.
- 157 Z. Babaei, A. N. Chermahini, M. Dinari, M. Saraji and A. Shahvar, *Sustainable Energy Fuels*, 2019, **3**, 1024–1032.
- 158 Z. Yang, P. Chen, W. Hao, Z. Xie, Y. Feng, G. Xing and L. Chen, *Chem. – Eur. J.*, 2021, **27**, 3817–3822.
- 159 J. Liu, H. Zhu, Z. Qiang, Q. Zhao, C. Que, C. Liu, M. Li, J. Cao, J. Wang and X. Yang, *Sep. Purif. Technol.*, 2025, **364**, 132466.
- 160 W. Meng, Q. Xue, J. Zhu and K. Zhang, *npj Clean Water*, 2024, **7**, 23.
- 161 Y. Yang, D. Wang, Q. Liu, Y. Zou, M. Wang, L. Li, Y. Lan, Y. Xiao, J. Hu, H. Zhang, M. Hu, K. Zhang, J. Wu and D. Gao, *Sep. Purif. Technol.*, 2025, **354**, 129280.
- 162 Z. Sun, X. Yang, X. Huang, M. Zhang, G. Bian, Y. Qi, X. Yang and W. Zhang, *New J. Chem.*, 2019, **43**, 16676–16684.
- 163 R. M. N. Kalla, M.-R. Kim and I. Kim, *Ind. Eng. Chem. Res.*, 2018, **57**, 11583–11591.
- 164 J. Li, F. Xu, Y. Chen, Y. Han and B. Lin, *ACS Appl. Energy Mater.*, 2022, **5**, 15856–15863.
- 165 Y. He, X. Lin, J. Chen and H. Zhan, *J. Membr. Sci.*, 2021, **635**, 119476.
- 166 J. Wolska, M. Frankowski, J. Jencyk and L. Wolski, *Sep. Purif. Technol.*, 2024, **343**, 127147.
- 167 K. Dong, J. Zhang, W. Luo, L. Su and Z. Huang, *Chem. Eng. J.*, 2018, **334**, 1055–1064.
- 168 J. Li, X. Wang, G. Chen, D. Li, Y. Zhou, X. Yang and J. Wang, *Appl. Catal., B*, 2015, **176–177**, 718–730.
- 169 Q. Sun, K. Hu, K. Leng, X. Yi, B. Aguila, Y. Sun, A. Zheng, X. Meng, S. Ma and F.-S. Xiao, *J. Mater. Chem. A*, 2018, **6**, 18712–18719.
- 170 X. Zhao, Q. Wang, R. Kunthom and H. Liu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 6657–6665.
- 171 B.-J. Chen, J.-P. Wu, Rukhsana, F.-Y. Xiang, Y.-M. Shen and X.-Z. Liang, *Kinet. Catal.*, 2024, **65**, 513–520.
- 172 C. Yadav, V. K. Maka, S. Payra and J. N. Moorthy, *ACS Appl. Polym. Mater.*, 2020, **2**, 3084–3093.
- 173 X. Wang, P. Ma, C. Yu, J. Bian and X. Meng, *Appl. Catal., A*, 2025, **700**, 120298.
- 174 N. Wei, Q. Wang, L. Guo, D. Wang, T. Chen and G. Wang, *ChemistrySelect*, 2024, **9**, e202304065.
- 175 Y. Li, L. Xing, Z. Chen, G. Zhan, B. Yuan, Y. Peng, L. Wang and J. Li, *Appl. Catal., B*, 2024, **358**, 124350.
- 176 Q. Xue, L. Cheng, Q. Qu, L. Yang, C. Fang, H. Li, J. Ding, H. Wan and G. Guan, *J. Environ. Chem. Eng.*, 2024, **12**, 113718.
- 177 Q. Yan, H. Xu, X. Jing, H. Hu, S. Wang, C. Zeng and Y. Gao, *RSC Adv.*, 2020, **10**, 17396–17403.
- 178 R. M. N. Kalla, S. S. Reddy and I. Kim, *Catal. Lett.*, 2019, **149**, 2696–2705.
- 179 L. A. Kulikov, M. A. Bazhenova, Y. S. Bolnykh, D. A. Makeeva, M. V. Terenina, Y. S. Kardasheva, A. L. Maximov and E. A. Karakhanov, *Pet. Chem.*, 2022, **62**, 1195–1203.
- 180 S. Daliran, S. F. Hosseini, A. R. Oveisi, E. Sanchooli, E. Rodríguez-Castellón and D. Ballesteros-Plata, *BMC Chem.*, 2025, **19**, 215.
- 181 G. D. Kotkar and S. G. Tilve, *ACS Omega*, 2024, **9**, 47543–47556.
- 182 S. S. Reddy, R. M. N. Kalla, A. Varyambath and I. Kim, *Catal. Commun.*, 2019, **126**, 15–20.
- 183 Z. Alishahi, M. Torabi, M. A. Zolfigol and M. Yarie, *J. Solid State Chem.*, 2023, **324**, 124119.
- 184 T. Mondal, J. Seth, S. Sarkar and S. M. Islam, *New J. Chem.*, 2023, **47**, 8885–8893.
- 185 S. Darvishi, S. Sadjadi and A. Rezvani, *J. Phys. Chem. Solids*, 2025, **199**, 112555.
- 186 M. Du, A. M. Agrawal, S. Chakraborty, S. J. Garibay, R. Limvorapitux, B. Choi, S. T. Madrahimov and S. T. Nguyen, *ACS Sustainable Chem. Eng.*, 2019, **7**, 8126–8135.
- 187 B. Ye, W. Zhang, R. Zhou, Y. Jiang, Z. Zhong and Z. Hou, *New J. Chem.*, 2022, **46**, 6756–6764.
- 188 J. Artz, I. Delidovich, M. Pilaski, J. Niemeier, B. M. Kübber, K. Rahimi and R. Palkovits, *RSC Adv.*, 2018, **8**, 22392–22401.
- 189 Y.-R. Du, B.-H. Xu, J.-S. Pan, Y.-W. Wu, X.-M. Peng, Y.-F. Wang and S.-J. Zhang, *Green Chem.*, 2019, **21**, 4792–4799.
- 190 D. Yuan, L. Li, F. Li, Y. Wang, F. Wang, N. Zhao and F. Xiao, *ChemSusChem*, 2019, **12**, 4986–4995.
- 191 W. Song, Y. Zhang, A. Varyambath, J. S. Kim and I. Kim, *Green Chem.*, 2020, **22**, 3572–3583.
- 192 Z. Guo and X. Liang, *Res. Chem. Intermed.*, 2024, **50**, 705–722.
- 193 P. Schweng, D. Guerin-Faucheur, F. Kleitz and R. T. Woodward, *Chem. Commun.*, 2024, **60**, 14395–14398.
- 194 J. Kim, S. Ravi, K. Kim, Y. Choi, H.-H. Park and Y.-S. Bae, *ACS Appl. Mater. Interfaces*, 2023, **15**, 48485–48494.
- 195 X. Jia, W. Xie and H. Li, *React. Chem. Eng.*, 2025, **10**, 1932–1948.
- 196 S. Y. Kang, Y. N. Lim, Y.-J. Cheong, S. M. Lee, H. J. Kim, Y.-J. Ko, B. Y. Lee, H.-Y. Jang and S. U. Son, *Ind. Eng. Chem. Res.*, 2017, **56**, 10235–10241.
- 197 W. Kang, C. Zhang, G. Li, P. Zhang and Y. Li, *Microporous Mesoporous Mater.*, 2022, **343**, 112151.



- 198 W. Kang, X. Wang, Y. Ren, P. Zhang, A. Huang and G. Li, *RSC Adv.*, 2024, **14**, 34578–34585.
- 199 K. Gong, C. Li, D. Zhang, H. Lu, Y. Wang, H. Li and H. Zhang, *Mol. Catal.*, 2022, **519**, 112139.
- 200 G. Kalita, N. Deka, D. Paul, L. Thapa, G. K. Dutta and P. N. Chatterjee, *Synlett*, 2020, 304–308.
- 201 A. A. Raza, S. Ravi, S. S. Tajudeen and A. K. I. Sheriff, *J. Solid State Chem.*, 2021, **302**, 122417.
- 202 Y. Pang, B. Wang, Y. Kang, J. Li, X. Gu, G. Kang, X. Yan, Y. Li and L. Chen, *Chem. Eng. Sci.*, 2022, **260**, 117933.
- 203 D. Jiang, X. Xu, Y. Bando, S. M. Alshehri, M. Eguchi, T. Asahi and Y. Yamauchi, *Bull. Chem. Soc. Jpn.*, 2024, **97**, uoae074.
- 204 L. Cui, H. Yuan, B. Yang, Q. Zhang, H. Zhang and J. Li, *Desalination*, 2025, **612**, 118971.
- 205 S.-S. Mortazavi, A. Abbasi and M. Masteri-Farahani, *Colloids Surf., A*, 2020, **599**, 124703.
- 206 F. Yang, W. Xue, D. Zhang, F. Li and Y. Wang, *React. Kinet., Mech. Catal.*, 2016, **117**, 329–339.
- 207 W. Xue, L. Sun, F. Yang, Z. Wang and F. Li, *Materials*, 2016, **9**, 833.
- 208 B. Hoque, M. Y. Khan, A. Hanif, M. Abdelnaby, A. Helal, A. Khan, M. Usman and Q. A. Drmosh, *J. Environ. Chem. Eng.*, 2024, **12**, 112695.
- 209 D. Ma, Y. Song, H. Zhao, C. Yu, Y. Zhang, C. Li and K. Liu, *ACS Sustainable Chem. Eng.*, 2023, **11**, 6183–6190.
- 210 Y.-R. Du, X. Yang, Y.-F. Wang, P.-X. Guan, R. Wang and B.-H. Xu, *Mol. Catal.*, 2022, **520**, 112164.
- 211 M. Yin, L. Wang and S. Tang, *ACS Appl. Mater. Interfaces*, 2022, **14**, 55674–55685.
- 212 H. Veisi, S. Taheri and S. Hemmati, *Green Chem.*, 2016, **18**, 6337–6348.
- 213 N. Iranpoor, H. Firouzabadi and E. E. Davan, *Tetrahedron Lett.*, 2013, **54**, 1813–1816.
- 214 L. A. Kulikov, M. A. Bazhenova, I. S. Bolnykh, A. L. Maximov and E. A. Karakhanov, *Catal. Lett.*, 2024, **154**, 6106–6122.
- 215 F. Doraghi, F. Mohaghegh, O. H. Qareaghaj, B. Larijani and M. Mahdavi, *RSC Adv.*, 2023, **13**, 13947–13970.
- 216 J. Chun, S. Padmanaban and Y. Lee, *J. Am. Chem. Soc.*, 2025, **147**, 16642–16652.
- 217 A. R. Oveisi, A. Khorramabadi-zad and S. Daliran, *RSC Adv.*, 2016, **6**, 1136–1142.
- 218 R. Ghorbani-Vaghei, A. Davood, S. Daliran and A. R. Oveisi, *RSC Adv.*, 2016, **6**, 29182–29189.
- 219 A. T. Khan, A. Ghosh and M. Musawwer Khan, *Tetrahedron Lett.*, 2012, **53**, 2622–2626.
- 220 N. Salehi and B. B. Fatameh Mirjalili, *RSC Adv.*, 2017, **7**, 30303–30309.
- 221 M. Masteri-Farahani, M. Ghahremani and M. Niakan, *Energy Fuels*, 2025, **39**, 15225–15241.
- 222 A. Priya, Y. Hu, J. Mou, C. Du, K. Wilson, R. Luque and C. S. K. Lin, in *Handbook of Biofuels Production*, ed. R. Luque, C. S. K. Lin, K. Wilson and C. Du, Woodhead Publishing, 3rd edn, 2023, ch. 1, pp. 3–24, DOI: [10.1016/B978-0-323-91193-1.00002-0](https://doi.org/10.1016/B978-0-323-91193-1.00002-0).
- 223 H. Fukuda, A. Kondo and H. Noda, *J. Biosci. Bioeng.*, 2001, **92**, 405–416.
- 224 M. Hong, J. Chen and E. Y. X. Chen, *Chem. Rev.*, 2018, **118**, 10551–10616.
- 225 X. Wang, D. Wu, X. Mu, W. Kang, G. Li, A. Huang and Y. Xie, *Polymers*, 2023, **15**, 555.
- 226 L.-p Yue, F.-x Kong, Y. Wang, G.-d Sun and J.-f Chen, *J. Membr. Sci.*, 2023, **685**, 121877.
- 227 S. Borsap, M. Charernsuk, K. Tangjituabun and J. Tantirungrotechai, *ChemCatChem*, 2025, **17**, e202500342.
- 228 D. Azarifar, R. Ghorbani-Vaghei, S. Daliran and A. R. Oveisi, *ChemCatChem*, 2017, **9**, 1992–2000.
- 229 E. Merino, E. Verde-Sesto, E. M. Maya, M. Iglesias, F. Sánchez and A. Corma, *Chem. Mater.*, 2013, **25**, 981–988.
- 230 Q. Liu, Z. Li, J. Sun, Y. Lan, J. Hu, Y. Xiao, F.-Q. Yang and D. Gao, *J. Sep. Sci.*, 2025, **48**, e70101.
- 231 L. Ma, M. M. Wanderley and W. Lin, *ACS Catal.*, 2011, **1**, 691–697.
- 232 K. N. Parida, C. Yadav, S. Jindal, P. Tamuly, Y. D. Vankar and J. N. Moorthy, *Chem. – Eur. J.*, 2025, **31**, e202404525.
- 233 X. Luo, S. Zhou, S. Zhou, D. Wang, Y. Zhang, X. Hao, G. Liu, P. Liu and P. Gu, *ACS Appl. Polym. Mater.*, 2024, **6**, 10615–10624.
- 234 N. Taheri and M. Dinari, *Appl. Surf. Sci.*, 2023, **18**, 100543.
- 235 L. Yang, Z. Zhan, L. Zhao, C. Zhang, S. Wang, W. Hu and G. Zhu, *Chem. Sci.*, 2025, **16**, 775–783.
- 236 Q. Li, S. Li, C. Yu, Z. Zhan, G. Cheng, B. Tan and S. Ren, *Commun. Mater.*, 2024, **5**, 255.
- 237 Y. Guo, Q.-M. Hasi, S. Hu, S. Jiang, J. Long, C. Xiao, Y. Zhang and L. Chen, *Langmuir*, 2024, **40**, 23382–23397.
- 238 Y. Zhang, X. Hu, K. Jiang, Z. Fan, D. Wang, P. Gu, S. Zhou and Z. Li, *Polymer*, 2024, **313**, 127738.
- 239 J. Wolska, M. Muńko, H. El Siblani, I. Telegeiev, M. Frankowski, A. Szwajca, J. Walkowiak-Kulikowska, M. El-Roz and L. Wolski, *J. Environ. Chem. Eng.*, 2023, **11**, 110429.
- 240 Y. Zhu, R. Ding, S. Chen, X. Qu, Y. Yang and X. Zhang, *Sep. Purif. Technol.*, 2024, **333**, 125857.
- 241 M. Bi, Y. Ma, J. Chai, Y. Su and Y. Xia, *Sep. Purif. Technol.*, 2025, **375**, 133810.
- 242 B. Li, J. Chen, J. Xiao, L. Zhao and H. Qiu, *ACS Appl. Nano Mater.*, 2023, **6**, 2498–2506.
- 243 S. Liang, Q. Meng, Z. Liu, H. Cai, H. Liu, M. Rong and L. Yang, *Chem. Eng. J.*, 2024, **499**, 155850.
- 244 Y. Xu, J. Wei and X. Chen, *Chemosensors*, 2022, **10**, 248.
- 245 M. Khosravani, M. Dehghani Ghanatghestani, F. Moeinpour and H. Parvaresh, *Heliyon*, 2024, **10**, e25423.
- 246 Y. Wang, L. Zhang, S. Wang, H. Feng, F. Pang, H. Li, B. Liu, X. Zang, S. Zhang and Z. Wang, *J. Water Process Eng.*, 2024, **68**, 106321.
- 247 Y. Guo, Q.-M. Hasi, J. Yu, Y. Guo, L. Song, S. Wu, X. Luo and L. Chen, *Sep. Purif. Technol.*, 2023, **324**, 124518.
- 248 W. Jin, J. Liu, N. Huang, Z. Wang, Y. Zhang, Y. Peng, C. Gong, Y. S. Ok and Z. Xu, *ACS ES&T Eng.*, 2023, **3**, 1511–1520.
- 249 D. W. Kang, M. Kang, M. Moon, H. Kim, S. Eom, J. H. Choe, W. R. Lee and C. S. Hong, *Chem. Sci.*, 2018, **9**, 6871–6877.



- 250 Z. Xuan, X. Sun, X. Zhang, G. Zhang, F. Wang and D. Liu, *Chem. Eng. J.*, 2025, **515**, 163958.
- 251 K. Zong, K. Jiang, X. Bao and D. Deng, *J. Environ. Chem. Eng.*, 2025, **13**, 116082.
- 252 S.-Q. Deng, Y.-L. Miao, Y.-L. Tan, H.-N. Fang, Y.-T. Li, X.-J. Mo, S.-L. Cai, J. Fan, W.-G. Zhang and S.-R. Zheng, *Inorg. Chem.*, 2019, **58**, 13979–13987.
- 253 Q. Liao, C. Ke, X. Huang, G. Zhang, Q. Zhang, Z. Zhang, Y. Zhang, Y. Liu, F. Ning and K. Xi, *J. Mater. Chem. A*, 2019, **7**, 18959–18970.
- 254 J. Abdi, M. Vossoughi, N. M. Mahmoodi and I. Alemzadeh, *Chem. Eng. J.*, 2017, **326**, 1145–1158.
- 255 N. A. Khan, H. J. An, D. K. Yoo and S. H. Jhung, *J. Hazard. Mater.*, 2018, **360**, 163–171.
- 256 X. Song, J. Lu, M. Liu, L. Tang, L. Sun, R. Jiang and L. Zhang, *Chem. Eng. J.*, 2024, **481**, 148714.
- 257 J. Missau, D. A. Bertuol and E. H. Tanabe, *Appl. Clay Sci.*, 2021, **214**, 106297.
- 258 D. Shetty, S. Boutros, T. Skorjanc, B. Garai, Z. Asfari, J. Raya and A. Trabolsi, *J. Mater. Chem. A*, 2020, **8**, 13942–13945.
- 259 S. Chanpee, N. Kaewtrakulchai, N. Khemasiri, A. Eiad-ua and P. Assawasaengrat, *Molecules*, 2022, **27**, 5309.
- 260 Y. Zhang, Y. Zhang, X. Luo, Q. Gao, Y. Liu, X. Wang, S. Zhou, D. Wang, P. Gu and Z. Li, *Sep. Purif. Technol.*, 2023, **327**, 124799.
- 261 W. Li, Z. Xie, S. Xue, H. Ye, M. Liu, W. Shi and Y. Liu, *Sep. Purif. Technol.*, 2021, **276**, 119435.
- 262 J. A. Cotruvo Jr., *ACS Cent. Sci.*, 2019, **5**, 1496–1506.
- 263 A. Javed, P. Palafox Gonzalez and V. Thangadurai, *ACS Appl. Mater. Interfaces*, 2023, **15**, 29674–29699.
- 264 Y. Nagao, *ChemElectroChem*, 2024, **11**, e202300846.
- 265 R. Sahoo, B. Pramanik and M. C. Das, *Adv. Energy Mater.*, 2025, **15**, 2501491.
- 266 D. E. Matkin, I. A. Starostina, M. B. Hanif and D. A. Medvedev, *J. Mater. Chem. A*, 2024, **12**, 25696–25714.
- 267 V. Joseph and A. Nagai, *RSC Adv.*, 2023, **13**, 30401–30419.
- 268 J. Si, S. Zhang, S. Zhao, J. Li, X. Feng and B. Wang, *Energy Mater. Adv.*, 2025, **6**, 0334.
- 269 U. Sen, H. Usta, O. Acar, M. Citir, A. Canlier, A. Bozkurt and A. Ata, *Macromol. Chem. Phys.*, 2015, **216**, 106–112.
- 270 B. Maria Mahimai, G. Sivasubramanian, K. Sekar, D. Kannaiyan and P. Deivanayagam, *Mater. Adv.*, 2022, **3**, 6085–6095.
- 271 M. Arun, S. Giddey, P. Joseph and D. S. Dhawale, *J. Mater. Chem. A*, 2025, **13**, 11236–11263.
- 272 A. Zucconi, J. Hack, R. Stocker, T. A. M. Suter, A. J. E. Rettie and D. J. L. Brett, *J. Mater. Chem. A*, 2024, **12**, 8014–8064.
- 273 Q. Li, J. O. Jensen, R. F. Savinell and N. J. Bjerrum, *Prog. Polym. Sci.*, 2009, **34**, 449–477.
- 274 U. Sen, S. Ünügür Çelik, A. Ata and A. Bozkurt, *Int. J. Hydrogen Energy*, 2008, **33**, 2808–2815.
- 275 C.-M. Chang, H.-Y. Li, J.-Y. Lai and Y.-L. Liu, *RSC Adv.*, 2013, **3**, 12895–12904.
- 276 S. M. Haile, D. A. Boysen, C. R. I. Chisholm and R. B. Merle, *Nature*, 2001, **410**, 910–913.
- 277 N. Agmon, *Chem. Phys. Lett.*, 1995, **244**, 456–462.
- 278 H. Xu, S. Tao and D. Jiang, *Nat. Mater.*, 2016, **15**, 722–726.
- 279 T. Ogawa, K. Kamiguchi, T. Tamaki, H. Imai and T. Yamaguchi, *Anal. Chem.*, 2014, **86**, 9362–9366.
- 280 Y. Lu, S. Zhou, C. Zhu, J. Zhou and X. Feng, *Chem. – Eur. J.*, 2025, **31**, e202501116.
- 281 D. W. Kang, M. Kang and C. S. Hong, *J. Mater. Chem. A*, 2020, **8**, 7474–7494.
- 282 D. W. Kang, K. S. Lim, K. J. Lee, J. H. Lee, W. R. Lee, J. H. Song, K. H. Yeom, J. Y. Kim and C. S. Hong, *Angew. Chem., Int. Ed.*, 2016, **55**, 16123–16126.
- 283 D. Zhang, Y. Gao, T.-X. Luan, K. Cheng, C. Li and P.-Z. Li, *Microporous Mesoporous Mater.*, 2021, **325**, 111348.
- 284 Z. Li, Y. Yao, D. Wang, M. M. Hasan, A. Suwansoontorn, H. Li, G. Du, Z. Liu and Y. Nagao, *Mater. Chem. Front.*, 2020, **4**, 2339–2345.
- 285 S. Y. Kim, M. Kang, D. W. Kang, H. Kim, J. H. Choe, H. Yun and C. S. Hong, *Angew. Chem., Int. Ed.*, 2023, **62**, e202214301.
- 286 S. F. Winterstein, A. F. Privalov, C. Greve, R. Siegel, B. Pötzschner, M. Bettermann, L. Adolph, J. Timm, R. Marschall, E. A. Rössler, E. M. Herzig, M. Vogel and J. Senker, *J. Am. Chem. Soc.*, 2023, **145**, 27563–27575.
- 287 S. F. Winterstein, M. Bettermann, J. Timm, R. Marschall and J. Senker, *Molecules*, 2024, **29**, 1669.
- 288 B. Tang, S.-T. Liu, X.-R. Ma, F.-J. Zhao, G.-Q. Zhang, X.-S. Gao, J. Yang, H.-B. Luo and X.-M. Ren, *ACS Appl. Mater. Interfaces*, 2025, **17**, 42004–42010.
- 289 H. Kim, I. Song, J. Kim, H. Seo, N. Park, H. J. Choi, W. Yang, Y.-E. Sung, W. B. Lee and J.-C. Lee, *ACS Appl. Mater. Interfaces*, 2025, **17**, 23155–23164.
- 290 Afzal, Y. Ren, S. Wang, H. Ma, S. Yuan, Q. Zhao, M. B. Wadud, X. Liang, Q. Pan, G. He and Z. Jiang, *J. Membr. Sci.*, 2025, **722**, 123863.
- 291 V. Joseph, K. Maegawa, M. Wlazło, M. J. Potrzebowski, K. Lyczko and A. Nagai, *Chem. Mater.*, 2025, **37**, 2561–2568.
- 292 Z. Yin, H. Geng, P. Yang, B. Shi, C. Fan, Q. Peng, H. Wu and Z. Jiang, *Int. J. Hydrogen Energy*, 2021, **46**, 26550–26559.
- 293 T. Xu, R. Yu, H. Bian and N. Zhang, *Inorg. Chem. Commun.*, 2025, **176**, 114348.
- 294 L. Zhang, Z. Gao, Y. Kong, N. Xing, X. Pang, Z. Liu, Z. Yao, S. Zhu, H. Wu and Z. Jiang, *J. Membr. Sci.*, 2024, **708**, 123088.
- 295 S. Moorthy, R. Sudhakaran, P. Dayalan and P. Deivanayagam, *J. Polym. Sci.*, 2025, **63**, 4153–4163.
- 296 B. He, Y. Lin, Q. Zhou and S. Tang, *Int. J. Hydrogen Energy*, 2024, **69**, 391–400.
- 297 D. Zhang, X. Li, L. Yang, L. Qian, G. Song, J. Guo and C. He, *Int. J. Hydrogen Energy*, 2025, **148**, 150089.
- 298 H. Zhao, B. Pang, F. Cui, W. Chen, G. Xie, X. Wu, R. Du, C. Liu and G. He, *J. Membr. Sci.*, 2025, **715**, 123479.
- 299 C.-X. Liu, S. Hwang, Y. Lee, Y. H. Ko, S. S. Park and E. Lee, *ACS Appl. Mater. Interfaces*, 2024, **16**, 48203–48210.
- 300 Y. Wu, Y. Wang, D. Zhang, F. Xu, L. Dai, K. Qu, H. Cao, Y. Xia, S. Li, K. Huang and Z. Xu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202313571.
- 301 X. Meng, Q. Peng, J. Wen, K. Song, L. Peng, T. Wu, C. Cong, H. Ye and Q. Zhou, *J. Appl. Polym. Sci.*, 2023, **140**, e53802.
- 302 B. Pang, G. Xie, H. Zhao, F. Cui, C. Liu, W. Chen, X. Yan, X. Wu and G. He, *Adv. Energy Mater.*, 2025, **15**, 2500523.

