



Superconductivity in topological Ψ -graphene

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Typical Dirac cones in graphene induce the absence of electronic density around the Fermi energy level (E_f), prohibiting intrinsic superconductivity. Here, we tested the theoretical superconducting properties of graphene after introducing pentagonal and heptagonal carbon rings into the structure. Generally, 5–7 polygons of a metastable Ψ -graphene monolayer break the hexagonal symmetry to form type-II Dirac cones by band crossings. The polyhedral structure maintains integrity under high temperatures. The large specific surface area of the Ψ -graphene monolayer facilitates the physical adsorption of NO molecules. Weak interactions of atomic bonding and antibonding features coexist with the Bader charge transfer in the carbon monolayer in close proximities to one another. The collective vibrations of carbon, nitrogen, and oxygen atoms provide good dynamic stability of the Ψ -graphene–NO adsorption system. In the Ψ -graphene monolayer, the shift of Dirac cones leads to the formation of visible Fermi surfaces, which motivates further investigation into their influences on the superconducting properties. Out-of-plane and in-plane carbon vibrations are attributed to phonon modes in mediation with electron couplings. After computing the Eliashberg function, we evaluated strong electron–phonon coupling, with the superconducting transition temperature reaching 22 K. These theoretical predictions can stimulate interests in exploring topological graphene allotropes of intrinsic superconductivity.

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Introduction

Superconductivity (SC) in graphene significantly broadens its application in electronic and quantum devices,^{1–4} with high carrier mobility and quantum Hall effect.^{5,6} However, the hexagonal honeycomb structure of mirror symmetry protects the Dirac-type band structure, prohibiting the emergence of SC.⁷ Twisted graphene bilayer superlattices with a “magic-angle” ($\approx 1.1^\circ$) produce robust SC at extremely low temperatures (< 3 K).^{8–10} Flat bands appear around zero Fermi energy, which results in half-filling correlated insulating states. Experimental observations of quantum oscillations confirm small Fermi surfaces under a perpendicular magnetic field of 0.4 T. Superconducting proximity effects enhance spin–orbit interactions in the heterostructure of single-layer WSe₂ and bernal bilayer graphene.^{11–13} Interestingly, the superconducting state of bilayer graphene can be observed even at zero magnetic field.

Beyond bilayer Moiré superlattices and bernal graphene, SC is found in ABC-stacked trilayer graphene.¹⁴ By tuning electrostatic gates under laser excitation, infrared-active phonon vibrations have demonstrated strong coupling strength with electrons. Another route to induce superconductivity in graphitic systems

involves electron doping *via* alkali or alkaline-earth intercalation (*e.g.*, Li, Na, K, Ca, and Sr), where reduced interlayer spacing enhances electron–phonon coupling (EPC).^{15–22} Larger deformation potentials with closer distances between doped metal atoms and the graphite plane (h) lead to stronger electron–phonon coupling (EPC). For example, the superconducting transition temperature (T_C) in bulk graphite intercalation compounds (GICs) increases from 1.65 K for SrC₆ ($h = 2.47$ Å) to 11.5 K for CaC₆ ($h = 2.26$ Å).^{16–20} A comparable h value of 2.32 Å for the GIC NaC₄ produces π -electron coupling with low-frequency vibrations, with an estimated T_C value of 41.2 K under a hydrostatic pressure of 5 GPa.²² However, quantum confinement between the carbon monolayer and adatom layers prevents π -electron occupation of the interlayer band and suppresses intrinsic SC in the GIC LiC₆ within the closest h distance of 1.85 Å.

Nevertheless, Li deposition on graphene removes the quantum confinement from bulk GICs. Li-covered graphene in single and double side constructs of LiC₆ and Li₂C₆ structures have T_C values of 8.1 and 18 K, respectively.²⁰ A downshift of the intercalant state to the Fermi energy level (E_f) is conducive to electron occupation for strong coupling between intercalant π electrons and carbon out-of-plane vibrations. Superconducting-gap measurements using angle-resolved photoemission spectroscopy provide further evidence regarding the superconducting enhancement of Li-decorated graphene LiC₆, with a T_C value of 5.9 K,²¹ which is close to the predicted value of 8.1 K. A tensile-strain level of 16.5% loaded for graphene is theoretically

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combined with a doping concentration of $4.65 \times 10^{14} \text{ cm}^{-2}$ for electrons and holes.²³ EPC is enhanced with enlarged Fermi surfaces to achieve a high T_C value approaching 30 K. Using diverse carbon rings as building blocks, various structural prediction methods such as AIRSS,²⁴ USPEX,^{25–27} CALYPSO,^{28–30} MAGUS,^{31–33} and RG2³⁴ have been applied to predict abundant graphene allotropes with sp^2 and sp^3 hybridizations.^{35–41} The characteristics of Dirac dispersions, such as tilt, Fermi-surface morphology, and nesting are reported to strongly affect the mode-resolved electron–phonon coupling and pairing interaction.^{42–44} These anisotropic electronic responses are of particular interest in carbon allotropes with distorted Dirac cones. Especially, anisotropic Dirac topology is exhibited in PHA,³⁸ SW,³⁹ POP,⁴⁰ and penta-graphene.⁴¹ Structure patterns specifically containing five-, seven-, and eight-fold rings break the hexagonal symmetry of a typical honeycomb structure. Robust metallicity emerges from delocalized π bands crossing over E_f .^{40,45,46} However, potential intrinsic SC combined with Dirac topology has been rarely investigated among the metallic graphene allotropes.

In this work, we intensively studied the electronic topology and intrinsic SC of the previously predicted psi (Ψ)-graphene structure.⁴⁵ We evaluated the topological Dirac cone features around E_f . Phonon dispersions demonstrated dynamic stability of Ψ -graphene under the physical adsorption of nitric oxide (NO) gas molecules. The electrons strongly coupled with carbon out-of-plane and in-plane vibrations, and Ψ -graphene demonstrated strong intrinsic SC with a T_C value of approximately 22 K.

Computational methods

We performed first-principles calculations using the Vienna *ab initio* simulation package (VASP)⁴⁷ based on density functional theory (DFT). We used the Perdew–Burke–Ernzerhof exchange–correlation functional (PBE)⁴⁸ under the generalized gradient approximation combined in VASP code. We set the total-energy cutoff to 600 eV, reaching a convergence level of 1×10^{-6} eV on each atom and a force of 5×10^{-3} eV \AA^{-1} per atom. We sampled a k mesh of $2\pi \times 0.03 \text{ \AA}^{-1}$ in the Brillouin zone (BZ). We accurately optimized the atom positions inner carbon plane, with the constant vacuum thickness reaching 25 \AA along the normal direction. By adding the D4 package^{49,50} to the PBE functional (PBE-D4), we calculated the adsorption energy regarding van der Waals (vdW) interactions between the carbon sheet and gas molecules. We utilized Lobster software⁵¹ to calculate the crystal orbital Hamilton population (COHP).⁵² Weak interactions between adjacent atomic orbitals were quantitatively analyzed for bonding and antibonding contributions. We used a Bayesian-learning algorithm employed in VASP to generate on-the-fly machine learning force fields (MLFF).^{53–56} We updated the MLFF from *ab initio* calculations when the Bayesian-error estimation exceeded a precise threshold for force error of 2×10^{-3} eV \AA^{-1} per atom. High-quality MLFF was employed in molecular dynamics simulations to discuss the stability of the large-scale structure under finite temperatures. We executed the tight-binding model from Wannier90

package⁵⁷ and WannierTools⁵⁸ to plot and analyze the topological band gap in the full BZ, respectively. By employing Quantum Espresso (QE) program,⁵⁹ electron–phonon coupling was computed by interpolation within the Monkhorst–Pack q grid of $(4 \times 1 \times 3)$ sampling over the BZ. We chose the norm-conserving pseudopotential for the carbon atom with a kinetic energy cutoff of 100 and 400 Rydberg (Ry) for wavefunctions and charge density, respectively. We calculated electronic Fermi surfaces within the eight-dense k grid of $(16 \times 4 \times 12)$. We set the iterative threshold of 1×10^{-14} for the self-consistent calculation. We further solved the Eliashberg function and obtained the electron phonon coupling coefficient (λ) and superconducting transition temperature T_C using the McMillan formula.⁶⁰

Results and discussion

Ψ -graphene sheets constructed with 5–6–7 carbon rings exhibited the space-group symmetry of *Pmma* (Number 51) with vacuum thickness of 25 \AA . Four Wyckoff-position (WP) sites were occupied by carbon atoms labeled as C1/C11 (equivalent atomic positions), C2/C3/C9/C10, C4/C6/C7/C12, and C5/C8 in Fig. S1 and Table S1 of the SI. The bond lengths between neighboring carbon atoms ranged from 1.409 to 1.503 \AA , with lattice constants of $a = 4.833 \text{ \AA}$ and $c = 6.705 \text{ \AA}$. The Ψ -graphene structure possessed a total energy of $-9.174 \text{ eV atom}^{-1}$, which was between the energies of Dirac cone bearing allotropes SW ($-9.191 \text{ eV atom}^{-1}$) and PHA graphene ($-9.133 \text{ eV atom}^{-1}$) (Table S2). The intermediate energy state for Ψ -graphene indicates possible thermodynamic meta-stability.

We investigated the structural stability of polygonal Ψ -graphene at high temperatures of 1500 K (Fig. 1) and 300 K (Fig. S2) through machine learning molecular dynamics simulations (MLMD). We extracted the averaged supercell of $(3 \times 1 \times 3)$ as the initial structure from the last 1 picosecond (ps) during *ab initio* molecular dynamics (AIMD) simulations for the NVT ensemble⁶¹ used for MLFF training. We further updated MLFF from AIMD using the NPT ensemble^{62,63} with a simulation time of 5 ps.⁶⁴ A large $(4 \times 1 \times 4)$ scale of the initial supercell system containing over 1700 carbon atoms was extended to employ efficient MLFF into MLMD with a total simulation time of 30 ps. Bayesian linear regression applied in MLMD samples a smoothed potential energy surface by learning the variances of the predictions from *ab initio* data. High-frequency thermal noise can be effectively suppressed with smaller temperature fluctuations. As shown in the left panels of Fig. 1(a–c), MLMD simulations achieved equilibrium states after 2 ps. Small fluctuations for temperature (T), pressure (P), and total energy (E) parameters reached a range of $\pm 100 \text{ K}$, $\pm 1 \text{ GPa}$, and $\pm 0.01 \text{ eV atom}^{-1}$, respectively, at around 1500 K and 0 GPa. We suggest that MLMD of the large-scale system exhibits greater accuracy for estimations of T , P , and E than AIMD [right panels in Fig. 1(a–c)]. The first two peaks of the pair distribution function $g(r)$ were located at carbon–carbon distances of 1.43 and 2.47 \AA , which indicated the closest and next-closest carbon atoms, respectively [red curve in Fig. 1(d)]. The neighboring carbon in

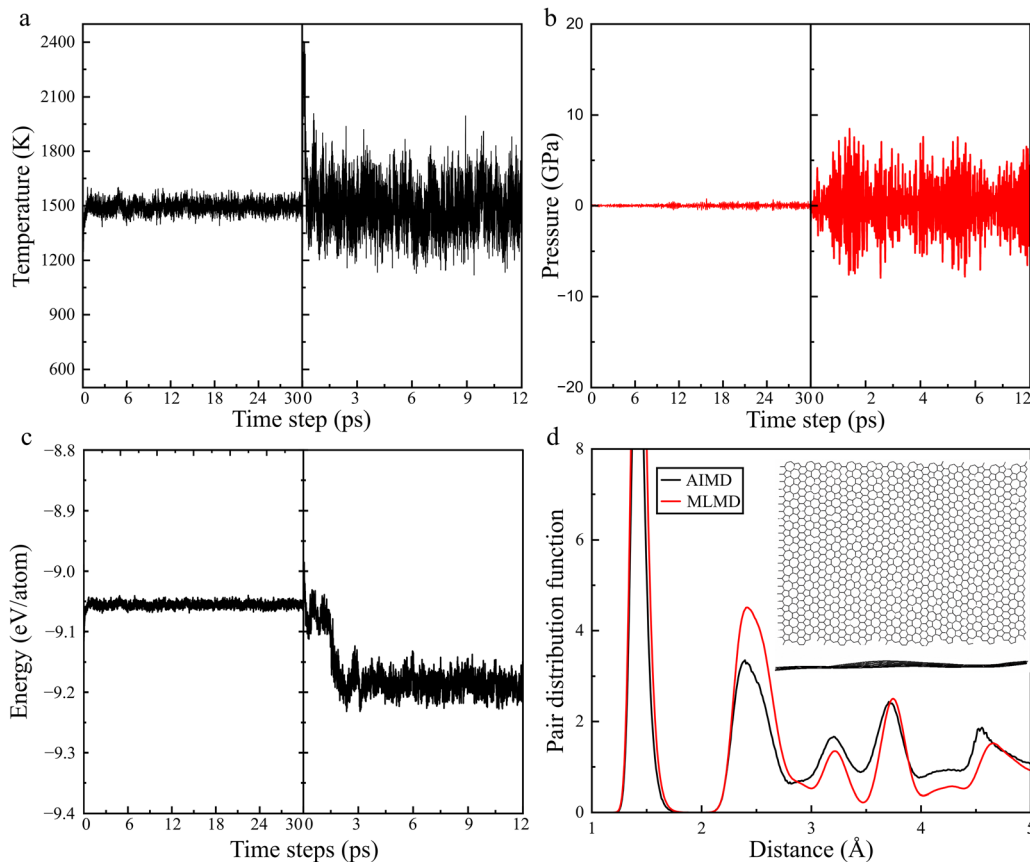


Fig. 1 The evolution of (a) temperature, (b) pressure, and (c) total energy during MLMD (30 ps, left panel) and AIMD (12 ps, right) simulations. (d) Pair distribution function $[g(r)]$ versus atomic distance at 1500 K. The inset graph is the averaged structure sketch obtained from the last 2 ps of MLMD.

the polyhedral structure was consistent with the pure AIMD NPT simulation (black curve). Furthermore, the averaged large-scale supercell structure confirms that Ψ -graphene maintains integrity at 1500 K. After cooling to room temperature (300 K), the structure remained stable, and the 5–6–7 ring configuration was well preserved (Fig. S2). Efficient MLMD calculations illustrate excellent structure stability for the Ψ -graphene monolayer under high temperature.

The Ψ -graphene sheet composed of 5–6–7 rings was predicted to have a larger specific surface area approaching $2.71 \text{ m}^2 \text{ mg}^{-1}$ than pristine graphene ($2.66 \text{ m}^2 \text{ mg}^{-1}$). The enlarged surface area potentially favors physical gas adsorption performance. To investigate the adsorption capability, we optimized the adsorption configurations by constructing a $(3 \times 1 \times 3)$ Ψ -graphene supercell with seven types of gas molecules. We took into consideration parallel (\parallel) and perpendicular (\perp) configurations for each molecule of H_2 , O_2 , N_2 , CO_2 , NO , NH_3 , and CO . Specifically, we used “H” to represent the center above the hollow site of carbon polygons and “C” to indicate the top site of WP carbon atoms. For instance, the adsorption site (AS) “H5” occupies the pentagonal center [Fig. S1(b)], while the AS “C1” represents the top site of WP C1 [Table S1 and Fig. S1(c)]. The initial distance (d) between the carbon monolayer and the molecule was set to 3.5 \AA to calculate physical adsorption [Fig. S3(a–c)] and further to 1.1 , 1.3 , and 1.5 \AA

[Fig. S3(d–f)] to compute “chemical adsorption”. We defined the adsorption energy (E_{ads}) as shown below: $E_{\text{ads}} = E_{\text{tot}} - E_{\text{mon}} - E_{\text{mol}}$. Here, E_{tot} , E_{mon} , and E_{mol} represent the total energy of the adsorption system, Ψ -graphene monolayer, and isolated molecule, respectively. We obtained negative E_{ads} for optimized adsorption states of Ψ -graphene-molecule systems by performing PBE-D4 calculations [Fig. 2(a)]. Among all molecules, we predicted that NO was the easiest to be adsorbed on Ψ -graphene sheet with the lowest E_{ads} reaching -0.687 eV . O_2 adsorption exhibited the next lowest E_{ads} of -0.549 eV . Detailed calculations at shorter distances ($d = 1.1$, 1.3 , and 1.5 \AA) illustrated consistent E_{ads} results for physical NO adsorption on Ψ -graphene, although there were slightly different ASs after in-plane optimization [Fig. S3(d–f)].

The adsorption of NO molecules especially affects the vibrations of neighboring carbon in Ψ -graphene. We computed the phonon dispersions [left panel in Fig. 2(b)] and phonon projected density of states (PDOS, right panel) for the Ψ -graphene-NO system in the AS “C4- \perp ” state. Compared with phonon dispersions of pristine Ψ -graphene (Fig. S4), carbon vibrations in the normal direction maintained the main contributions to phonon dispersions within a frequency range of $400\text{--}830 \text{ cm}^{-1}$. However, the oxygen (O) atom mainly contributed to flat acoustic branches at low-frequency around 35 cm^{-1} [red curve in Fig. 2(b)]. Vibrations of nitrogen (N) atoms in the normal

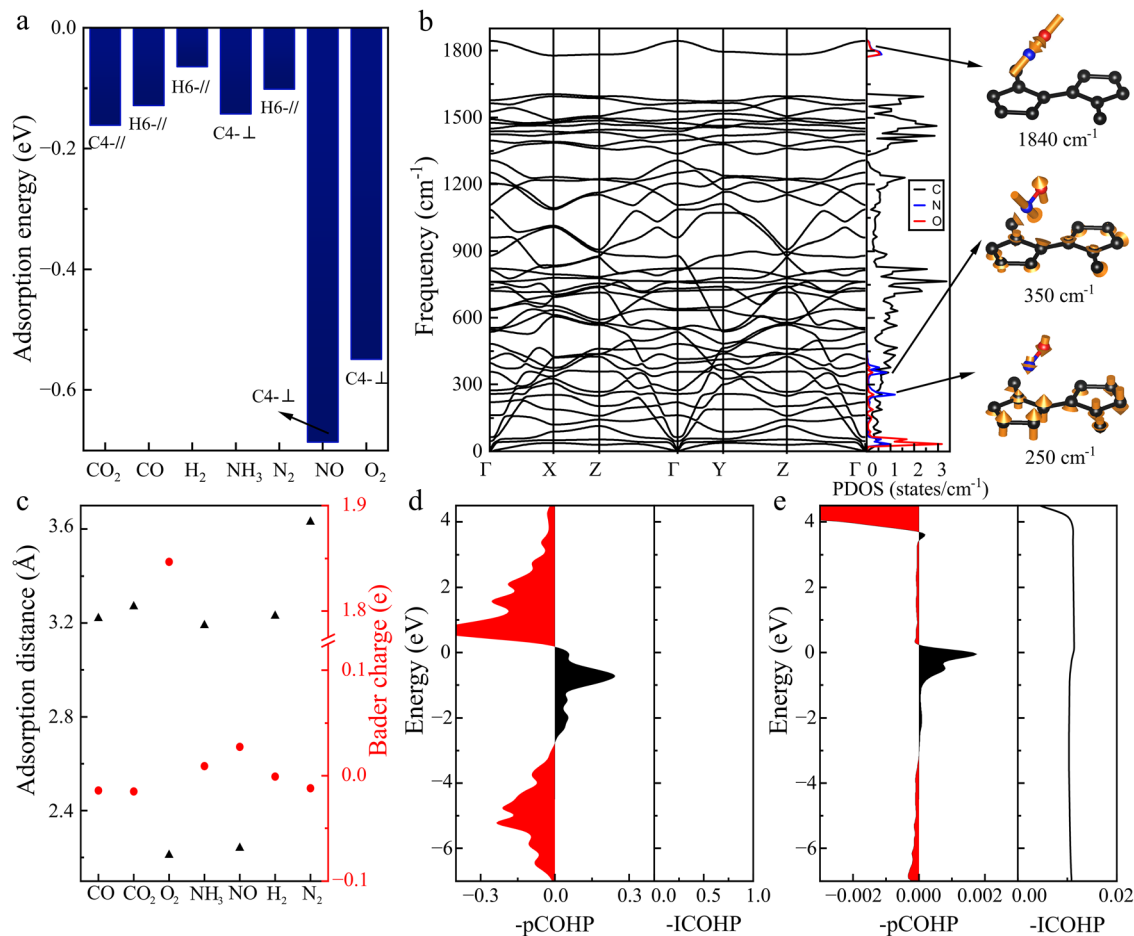


Fig. 2 (a) Adsorption energy by structure optimization for an initial distance of 3.5 Å. (b) Calculated phonon dispersion (left panel) and phonon density of states (PDOS, right) for NO adsorption of Ψ-graphene. (c) Bader charge transfer and adsorption distance between Ψ-graphene and adsorbed gas molecules. -pCOHP and -ICOHP results for Ψ-graphene-NO adsorption with a *d* of around (d) 2.2 Å between WP C4 and N, and (e) stretching *d* to 4.4 Å.

direction participated in phonon optical branches around 250 and 350 cm⁻¹, as shown in the blue curve in Fig. 2(b). The opposing vibration along the N=O bond resulted in the highest frequency of phonon mode, which approached 1840 cm⁻¹ (red). The in-plane carbon vibrations from WP C1 and C2 atoms were suppressed below a frequency of 1600 cm⁻¹ (Fig. S4). These calculations indicate that monolayer Ψ-graphene exhibits superior adsorption performance for the NO molecule, which is accompanied with significant phonon softening and hardening effects.

As shown in Fig. 2(c), Ψ-graphene adsorption of O₂ or NO molecules at low total energy takes advantage of the close distance *d* (black triangles) for Bader charge transfer (red solid dots). Specifically, almost 1.85 and 0.03e of Bader charge are transferred from O₂ and NO molecules, respectively, to C2, with *d* reaching 2.24 Å for NO. In particular, there are weak atomic interactions between neighboring C and N in the adsorption process. Bonding and antibonding from carbon 2p orbitals were quantitatively calculated by the projected COHP (pCOHP) and integral COHP (ICOHP) [Fig. 2(d)]. The vdW interactions mainly originate from the C bonding state of the pristine Ψ-graphene monolayer in an energy range of -2.8 eV below *E_F*,

with the -ICOHP peak reaching 0.79 eV [black area in Fig. 2(d)]. The N p orbital was attributed to the antibonding state in an energy range of 3.0 eV above the *E_F*. By doubly stretching the distance to 4.4 Å between WP C2 and the N atom [Fig. 2(e)], the positive -pCOHP values decreased (black) and the negative values disappeared (red) in an energy range between -7.0 to 4.5 eV. Decrement of the -ICOHP value indicates weaker atomic interactions for bonding and antibonding states. We suggest that this graphene allotrope serves as an excellent sensor for NO gas, with weak interactions in close proximity to Bader charge transfer.

The adsorption event with non-covalent interactions in the charge transfer distance perturbs the electronic band structures.⁶⁵ To further explore the superconducting properties after NO adsorption, we calculated the band structure along the high-symmetry line (HSL) of Γ-X-Z-Γ-Y-Z-Γ and partial density of states (pDOS) for NO adsorption (Fig. 3) onto pristine Ψ-graphene (Fig. S5). Upon NO physisorption in close proximity to the WP C4 site for carbon atom, pDOS revealed that the nitrogen and oxygen atoms contributed to the band dispersions in an energy-level range of ±1 eV. New Fermi surfaces open around Γ-Y and X-Z directions with elevated *E_F* as shown in the inset graph in Fig. 3.

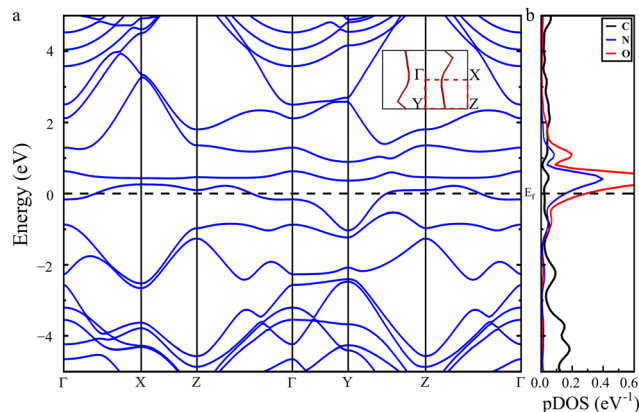


Fig. 3 (a) Band structure and (b) partial density of states (pDOS) for NO adsorption of Ψ -graphene. The inset graph in (a) is the Fermi surface.

Consequently, band dispersions attributed to C4 p electrons are repelled to be flattened around 1 eV above E_f along Z- Γ -Y-Z- Γ for the NO- Ψ -graphene system. Nevertheless, we suggest that the adsorption system still characterizes potential superconductivity with new Fermi surfaces under band regulations by NO physical adsorption.

We observed an oblique band crossing along the HSL Y-Z at energy levels of about 0.2 eV below E_f [Fig. S5(a)]. The isoenergy surface at the band-crossing point (down) showed high similarity with the Fermi surface (up), as shown in the insets of Fig. S5(a). We suggest that Dirac electrons participate in the formation of electron (red) and hole-pocket (green) types of intrinsic Fermi surfaces. The tight-binding models were constructed by s and p orbitals of C atoms using the Wannier90 package. The band crossing along HSL Y-Z particularly suggests potential topological semimetal features in the Ψ -graphene monolayer. We verified six type-II Dirac cones with the π Berry phase around the Y point marked as red and blue dots in Fig. 4(a). Two Dirac cones (marked by red dots) located at HSL Y-Z were connected with each other by mirror symmetry M_y of the same energy. Four Dirac cones of blue dots were connected by M_y and glide mirror symmetry G_x . After plotting band gaps in the BZ using WannierTools [Fig. 4(b)], there were no other band crossings in the full BZ. However, mediation with the adsorbed NO molecule breaks

the original mirror symmetry in Ψ -graphene, which leads to the disappearance of the band crossing along the Y-Z path [Fig. 3(a)].

Unique Fermi surfaces inspire us to explore EPC SC in the metallic Ψ -graphene monolayer. After solving the Eliashberg function of $\alpha^2F(\omega)$, we estimated the EPC parameter λ of 1.32 and logarithmic average frequency ω_{\log} of 218.3 K [Fig. 5(a)] using the following formulas:

$$\lambda = 2 \int d\omega \alpha^2 F(\omega) / \omega \quad \text{and}$$

$$\omega_{\log} = \exp \left\{ \lambda / 2 \left[\int d\omega \alpha^2 F(\omega) \log \omega / \omega \right] \right\}.$$

The collective vibrations of WP C2 and C5 carbon atoms in the normal direction were attributed to the peaks of the $\alpha^2F(\omega)$ function (red curve) at phonon frequency ω of 220 and 415 cm^{-1} , respectively (inset graphs). The ω_{\log} value was quickly enhanced to reach about 11.5% and 26.1% of the total value with increasing phonon frequency (blue curve). A high $\omega > 1470 \text{ cm}^{-1}$ contributed to 16.6% of the total ω_{\log} , which originated from WP C1 in-plane vibrations for the $\alpha^2F(\omega)$ peak. Phonon modes of low frequency ($\omega < 220 \text{ cm}^{-1}$) contributed to 68.2% of the total λ (green curve). We conducted T_C estimation using the conventional McMillan formula:

$$T_C = (\omega_{\log}/1.2) \exp[-1.04(1 + \lambda)/(\lambda - \mu^* - 0.62\lambda\mu^*)].$$

We systematically tested different σ values for gaussian broadening from 0.02 to 0.60 Ry with the Coulomb pseudopotential μ^* from 0.08 to 0.16 to reach the convergence of T_C . Fig. 5(b) clearly shows a convergent T_C value at a σ value of 0.60 Ry under considerable μ^* values. Therefore, we used $\mu^* = 0.10$ and $\sigma = 0.60$ Ry with the Methfessel-Paxton smearing scheme. The T_C value (black) increases with ω , which is consistent with the ω_{\log} curve (blue). Out-of-plane phonon vibrations of low frequency and in-plane modes of high frequency in the Ψ -graphene monolayer were suggested to mediate electron couplings. We propose that Ψ -graphene has a strong intrinsic SC EPC with a T_C value approaching 22 K, which is comparable to that of the predicted metastable PHH graphene (20 K).³⁹ However, the broken symmetry of the system for NO-adsorbed Ψ -graphene makes the evaluation

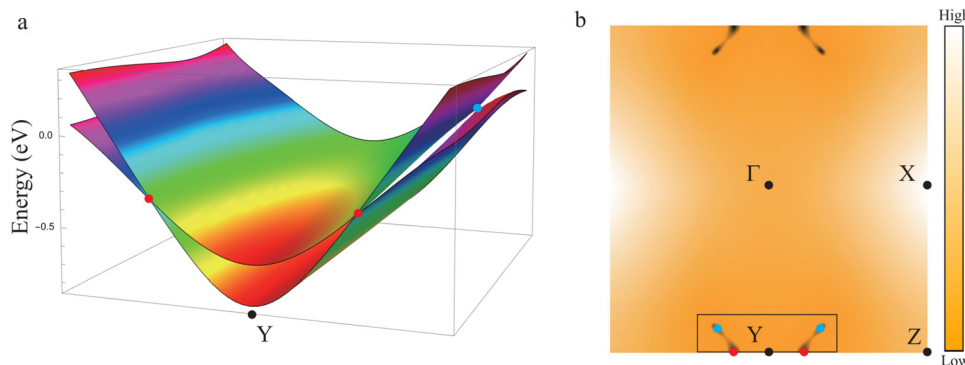


Fig. 4 (a) Energy dispersion around the Y point. The Dirac cones are marked by red and blue dots. (b) Band gap distribution in the full BZ, with the energy band dispersions around Y shown in the inset.

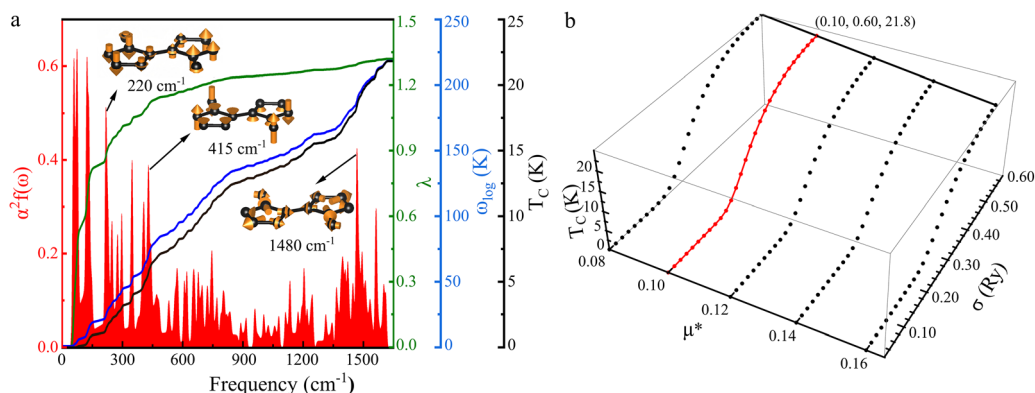


Fig. 5 (a) Calculations for EPC superconductivity, including the Eliashberg function $\alpha^2F(\omega)$, EPC (λ), logarithmic average frequency (ω_{\log}), and superconducting transition temperature (T_C). (b) T_C value as a function of the Coulomb pseudopotential μ^* and the parameter σ . The black dots denote the calculated data points, the red solid curve highlights the $\mu^* = 0.10$ across different σ , and the black solid line connects the data points at fixed $\sigma = 0.60$ across different μ^* .

of quantities [λ and $\alpha^2F(\omega)$] extremely time-consuming by performing QE code. We plan to further investigate this aspect in subsequent work.

Conclusions

Pentagonal and heptagonal carbon rings in Ψ -graphene break the hexagonal symmetry to bring about the emergence of electronic density of states at E_f . The 5–6–7 polyhedral structure remains intact under high temperature up to 1500 K during machine learning molecular dynamics simulations. The large specific surface area potentially facilitates the adsorption of gas molecules. In particular, single-layer Ψ -graphene exhibits the lowest adsorption energy of -0.687 eV for NO. Bader charge transfer occurs at the shortest distance around 2.2 Å. The Ψ -graphene–NO adsorption system possesses dynamic stability that involves weak chemical interactions based on COHP analyses. Band crossings below E_f originate in type-II Dirac cones, illustrating the topological semi-metallicity in Ψ -graphene. Shifted Dirac cones induce visible Fermi surfaces to encourage electron couplings with phonon mediation. We suggest that out-of-plane and in-plane carbon vibrations contribute to the low-frequency (220 and 415 cm^{-1}) and high-frequency (1470 cm^{-1}) phonon modes. We estimate that the electron–phonon coupling constant increases up to 1.32 with a logarithmic average frequency of 218.3 K. The conventional superconducting transition temperature approaches a peak value of 22 K. Therefore, Ψ -graphene possesses intrinsic SC with topological semi-metallicity.

Author contributions

Xiaolong Yu: writing – original draft, methodology, investigation, formal analysis, data curation, conceptualization. Zhongyan Lu: methodology, data curation, Zhaopeng Guo: writing – original draft, methodology, visualization, formal analysis, conceptualization. Kang Xia: writing – review & editing, methodology, visualization, validation, supervision, resources, conceptualization.

Conflicts of interest

There are no conflicts to declare.

Data availability

The data supporting this article describes lattice parameter (Table S1), total energy (Table S2), crystal graphic (Fig. S1), MD simulation at 300 K (Fig. S2), adsorption-energy calculation details (Fig. S3), phonon dispersions and PDOS (Fig. S4), and electronic band structure and pDOS (Fig. S5). The supplementary data have been included as part of the supplementary information (SI). Supplementary information is available. See DOI: <https://doi.org/10.1039/d5cp04499e>.

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