

ChemComm

Chemical Communications

Accepted Manuscript

This article can be cited before page numbers have been issued, to do this please use: A. T. Brennan, M. T. Pelly, A. S. Gibbs and A. W. Rost, *Chem. Commun.*, 2026, DOI: 10.1039/D5CC06230F.



This is an Accepted Manuscript, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about Accepted Manuscripts in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this Accepted Manuscript or any consequences arising from the use of any information it contains.

Journal Name

ARTICLE TYPE

Cite this: DOI: 00.0000/xxxxxxxxxx

Unconventional Magnetism and Residual Entropy in $\text{La}_7\text{Os}_3\text{O}_{18}^\dagger$ Ashley T. Brennan^{a,b}, Maximilian T. Pelly^{a,b}, Alexandra S. Gibbs,^{*a} and Andreas W. Rost^b

Received Date

Accepted Date

DOI: 00.0000/xxxxxxxxxx

In this communication we describe the crystal structure and unusual magnetic properties of $\text{La}_7\text{Os}_3\text{O}_{18}$, a new layered osmate with structurally discrete (non-corner-sharing) octahedra containing Os^{5+} with $S = \frac{3}{2}$ and strong spin-orbit coupling. Investigations through powder X-ray diffraction, magnetometry and heat capacity measurements demonstrate that the material undergoes an antiferromagnetic ordering transition at $T_N = 43 \pm 1$ K, with a further broad overturn in susceptibility at $T \approx 14$ K indicating low dimensional short range magnetic ordering. Curie-Weiss fits give an effective moment of $\mu_{eff} = 3.26 \mu_B$ per Os^{5+} and Weiss constant of $\theta_W \approx -92$ K. Heat capacity data reveal that it is the low temperature feature with which the majority of the magnetic entropy is associated. However between our lowest temperature of 2 K and well above T_N we find only approximately half of the expected magnetic entropy to be released. Magnetisation versus field measurements at 2 K reveal a hysteretic metamagnetic transition above 5 T further indicating that the magnetism below T_N is non-trivial in nature.

In comparison to 3d transition metal oxides, 5d compounds offer access to more spatially extended d orbitals and greater spin orbit coupling, which scales with atomic number Z between Z^2 and Z^4 ^{1,2}. These considerations frequently lead to physical properties that differ substantially from related compounds of the lighter transition metals, driving substantial research interest in the field.

Of the 5d metals, the osmates have remained relatively understudied due to the practical implications of the toxicity of OsO_4 . However, discoveries of polar metallicity in LiOsO_3 ³, charge ordering in $\text{Sr}_5\text{Os}_3\text{O}_{13}$ ⁴ and superconductivity in the β -pyrochlores AOs_2O_6 ($A = \text{K}, \text{Rb}, \text{Cs}$)^{5–7} have demonstrated the diversity of thermodynamic phases accessible through osmium chemistry.

Such states are supported by the variety of structures accessible when multiple oxidation states (+4, +5, +6, +7, +8) can be stabilised, and even mixed.

Amongst the osmates described in the literature, there are few with isolated OsO_6 octahedra. Those which have been studied, such as Li_5OsO_6 and Ca_3OsO_6 , tend to occupy the higher +6 and +7 Os oxidation states and thus have a smaller effective moment on each Os site. In this study we present the basic structural, magnetic and thermodynamic properties of the compound $\text{La}_7\text{Os}_3\text{O}_{18}$, which has isolated octahedrally coordinated Os^{5+} sites within a complex layered crystal structure.

Initially $\text{La}_7\text{Os}_3\text{O}_{18}$ was identified as an impurity during the synthesis of polycrystalline samples of other lanthanum osmates, but was subsequently isolated, and could be made using several different sets of conditions. *Note that caution is required when handling osmium and its compounds because highly toxic and volatile OsO_4 is produced at elevated temperatures and under oxidising conditions.* All handling of osmium and its compounds was carried out within an argon glove box or in a manner suited to air sensitive samples. The starting reagents (OsO_2 , min 83% Os, and La_2O_3 , $\geq 99.9\%$) were ground together and pressed into a pellet which was then sealed under vacuum in a quartz ampoule, along with a small quantity of AgO as an oxidising agent. AgO decomposes in two steps at approximately 200 and 400 °C to yield O_2 , and was kept separate from the pelleted reagents by encasing it in Pt foil. Significant excesses of OsO_2 were used (up to 5 times the stoichiometric quantity in the most extreme case) in order to counteract the loss of Os to OsO_4 which occurs readily at temperatures in excess of 500 °C. The sealed ampoules were heated in a tube furnace at temperatures between 800 and 950 °C for times ranging from 4 to 48 hours. Full synthesis details for each sample can be found in the Supplementary Information.[†]

The structure of $\text{La}_7\text{Os}_3\text{O}_{18}$ was identified by comparison with powder diffraction data from $\text{La}_7\text{Ru}_3\text{O}_{18}$ ⁹, which adopts space group $R\bar{3}c$. The related structure type of $\text{Sr}_5\text{Re}_2\text{O}_{12}$ ¹⁰ was found to yield a worse fit to the data, with La occupancies on the additional site refining to zero. Detailed structural analysis was

^a EaStCHEM, School of Chemistry, University of St Andrews, North Haugh, St Andrews, KY16 9ST, UK.

^b School of Physics and Astronomy, University of St Andrews, St Andrews, KY16 9SS, UK.

* E-mail: a.gibbs@st-andrews.ac.uk

[†] Supplementary Information available: [details of any supplementary information available should be included here]. See DOI: 00.0000/00000000.



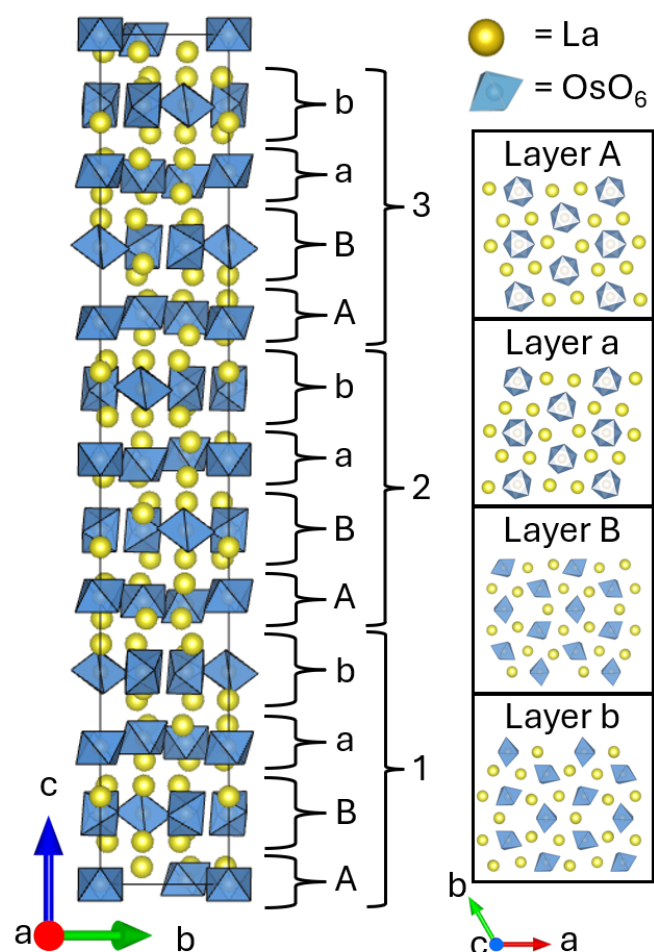


Fig. 1 A view of the structure of $\text{La}_7\text{Os}_3\text{O}_{18}$ along the a -axis, with unit cell in black, OsO_6 octahedra in blue and La in yellow. There are two principal layer types, each with two arrangements, pictured individually along the c -axis on the right. The ABab stacking sequence emerges when considering the structure in its entirety, and relates to the double hexagonal close packed ABAC sequence from osmium positions alone. Visualised using VESTA⁸.

carried out using Mo $K\alpha$ powder X-ray diffraction measured in Debye-Scherrer mode at room temperature with the sample sealed in a 0.3 mm diameter quartz capillary. No absorption correction was applied. A good fit to the data could be achieved via Rietveld refinement (see Supplementary Information[†]), giving $a = 9.8639(5)$ Å, $c = 56.3934(19)$ Å, $R_{wp} = 0.05412$, $\chi^2 = 2.07$ (see Supplementary Figure S1[†]), with little change in structural parameters from those of the ruthenate analogue. The bond valence sum gives 4.91, 4.54 and 4.42 for the three symmetry-inequivalent osmium sites, consistent with a 5+ oxidation state.¹¹ An impurity was identified as LaClO , with a refined phase fraction of 3.9 % for a representative sample. The Cl in this phase was found to come from an impurity in the OsO_2 starting reagent. LaClO is not magnetic and its contribution is negligibly small to the results we report here.

The structure adopted by $\text{La}_7\text{Os}_3\text{O}_{18}$ is quite intricate with an unusually large unit cell (see Figure 1). Considering only the Os positions, a distorted double hexagonal close packed structure

emerges. The double hexagonal close packing sequence is ABAC, however in $\text{La}_7\text{Os}_3\text{O}_{18}$, a distortion is applied to the 'A' layer, displacing the Os ions slightly from the plane. The resulting undulations produce a far more complex stacking sequence that expands the unit cell by $\sqrt{3}a$ and $3c$. This subcell-supercell relationship was previously observed in precession images from single crystal X-ray diffraction data by Mons and colleagues, working on the related structures of $\text{Sr}_5\text{Re}_2\text{O}_{12}$ and $\text{Ca}_5\text{Re}_2\text{O}_{12}$ ¹⁰.

By considering the OsO_6 octahedra, it becomes clear that there are two principal types of layer in the material; one makes up the undulating layer 'A' of the distorted double hexagonal close packed structure, the other making up both layers 'B' and 'C'. Layer 'A' has OsO_6 octahedra arranged with faces aligned with the ab plane, this layer having two of the three crystallographically inequivalent Os sites (panels A and a in Figure 1). The remaining Os site is located in the other layer type and has OsO_6 arranged without coplanar faces (panels B and b in Figure 1).

Lanthanum atoms are present in both layer types, located above and below the Os sites. Incorporating the lanthanum positions into the description of the structure leads to a more complete, though more complex, picture. The two layer types stack alternately, however every second occurrence of each layer has a transformation applied, leading to an ABab stacking sequence where layer a is equivalent to layer A with a twofold rotation about the a axis and layer b is equivalent to layer B after inversion through the centre of the unit cell. Layers B and b make up the B and C positions of the double hexagonal ABAC stacking of Os sites described above. The unit cell contains three of these stacks of four layers with the ABab sequence, each stack of four translated by $1/3$ in a and $-1/3$ in b .

Layer A/a (see Figure 1) has a La:Os ratio of 2:1 and is thinner than layer B/b which has a La:Os ratio of $2\frac{2}{3}$:1. There are three crystallographically distinct La positions, two occurring in the relatively La rich layer B/b and one in the relatively La poor layer A/a. Aside from the ruthenate analogue other examples of this structure type exist, though it is relatively rare^{12–14}. Recently tungstates in this class have garnered interest within the luminescent materials community for their separation of WO_6 octahedra and stability to dopants^{15–17}.

Magnetic properties were investigated under an applied field $\mu_0 H = 1$ T (Figure 2) using the VSM options for the Quantum Design MPMS 3 and PPMS. The behaviour was consistent across samples synthesised under varying conditions. The susceptibility, M/H , of $\text{La}_7\text{Os}_3\text{O}_{18}$ has paramagnetic behaviour at high temperatures, down to a transition temperature of $T_N = 43 \pm 1$ K (with the uncertainty indicating the degree of variation between different samples), where there is a maximum consistent with antiferromagnetic ordering. A second maximum is also observed centred on $T \approx 14$ K. The field cooled and zero field cooled data diverge below the higher temperature transition. These three principal features of the susceptibility were quantitatively and qualitatively reproducible across samples of varying quality, indicating that they are intrinsic, which suggests further complexity in the magnetic structure beyond simple collinear antiferromagnetism. Further measurements at low applied fields ($\mu_0 H = 0.01$ T, see Supplementary Figure S2[†]) show an additional feature at $T = 37$



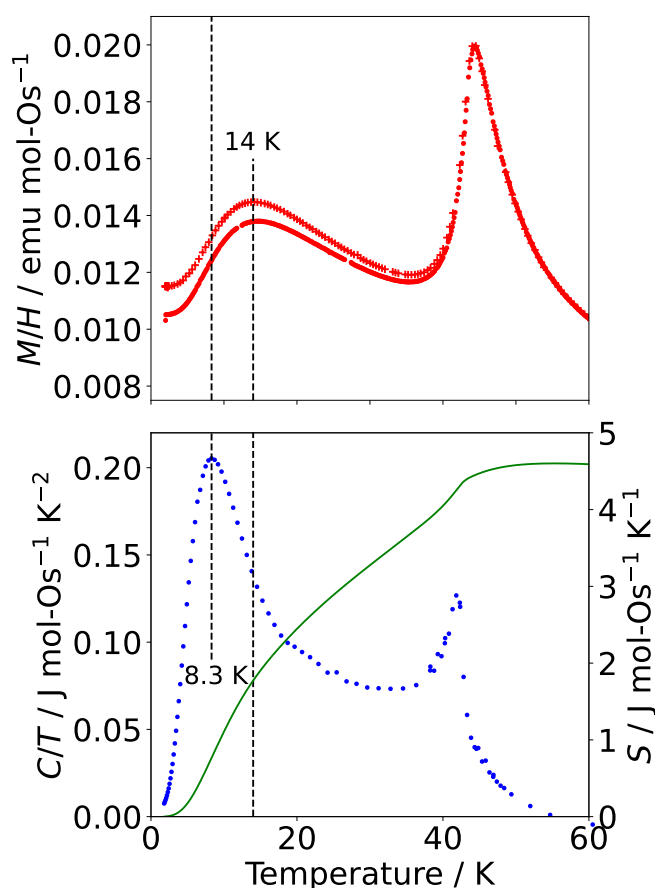


Fig. 2 Upper panel: Polycrystalline magnetic susceptibility data, for an applied field $\mu_0 H = 1$ T. Dots indicate zero field cooled and crosses indicate field cooled data. Lower panel: Heat capacity divided by temperature for the same temperature interval, with the integrated entropy plotted on the right axis.

K and a greater divergence between the zero field cooled and field cooled data. These features at low fields were also reproducible (see Supplementary Information[†]).

A Curie-Weiss fit of further variable temperature magnetic studies carried out at $\mu_0 H = 1$ T, with inclusion of a temperature-independent Van Vleck term and a core diamagnetism correction of $-144 \mu\text{emu mol-Os}^{-1}$ (Supplementary Figure S3[†]), yields $\theta \approx -92$ K, indicating relatively strong antiferromagnetic interactions, and moderate but not remarkable frustration. The effective moment is $3.26 \mu_B$ per Os^{5+} , lower than the expected spin only value of $3.87 \mu_B$ for $S = \frac{3}{2}$, likely the result of spin-orbit coupling, and the Van Vleck contribution $357 \pm 3 \mu\text{emu mol-Os}^{-1}$. Measurements of magnetisation versus field (Figure 3) reveal a metamagnetic transition in high field with hysteresis, resulting in a double hysteresis loop. In addition, a smaller hysteresis loop is also present at lower fields at 10 K (see inset, Figure 3).

In the case of $\text{La}_7\text{Os}_3\text{O}_{18}$, there is an isostructural ruthenate analogue, $\text{La}_7\text{Ru}_3\text{O}_{18}$ ^{9,19}, which is also a spin- $\frac{3}{2}$ system with $\mu_{\text{eff}} = 3.49 \mu_B/\text{Ru}^{5+}$, and $\theta_W = -58$ K. The ruthenate was found to undergo an antiferromagnetic ordering transition at $T_N = 14$ K, significantly lower than in $\text{La}_7\text{Os}_3\text{O}_{18}$ and the authors observed field dependence, with the transition temperature suppressed as

applied field was increased.

To investigate the intriguing findings of the susceptibility measurements further, and to confirm that these features arise intrinsically from $\text{La}_7\text{Os}_3\text{O}_{18}$, zero field heat capacity measurements were conducted on two polycrystalline samples of $\text{La}_7\text{Os}_3\text{O}_{18}$ from 303 K to 1.8 K using the heat capacity option of a Quantum Design PPMS. These data were fitted between 49.4 and 303 K with a Debye-Einstein model in order to estimate and subtract the phonon contribution to the specific heat. It should be noted that the subtraction of phonon specific heat in the absence of a non-magnetic isostructural phononic reference sample is not trivial. Our data is well described by a roughly equal combination of one Debye and one Einstein mode with characteristic temperatures $\theta_D = (262 \pm 3)$ K and $T_E = (586 \pm 7)$ K respectively, with the remaining optical modes at too high an energy to contribute. We note that this widely used model does violate the total phonon mode count and we discuss alternatives for modelling the phonon specific heat in the Supplementary Information[†]. The resulting magnetic contribution to the heat capacity is shown in Figure 2. The data show a sharp peak at 42 K and a broad crossover centred around 8.3 K, with a plateau between the two features. The peak at 42 K corresponds to T_N from the susceptibility data, while the crossover centred on 8.3 K is associated with the temperature where there is the maximum change in the susceptibility as the temperature drops below the broad peak at 14 K. This crossover, with its characteristic behaviour in susceptibility and heat capacity, is potentially indicative of low-dimensionality driven short-range ordering²⁰. The small peak in low-field χ at $T = 37$ K is located in the plateau region of the magnetic heat capacity data.

From the C/T data, it becomes clear that the majority of the magnetic entropy is associated with the lower temperature, broad crossover. Integration yielded a magnetic contribution to the entropy of the system which approaches $S = 4.6 \text{ J mol}^{-1} \text{ K}^{-1}$ for both peaks. The expected value for a $S = \frac{3}{2}$ system calculated by $R \ln(2S + 1)$ is $11.5 \text{ J mol}^{-1} \text{ K}^{-1}$, more than double what we observe, indicating the potential for incomplete magnetic ordering within this system (see Supplementary Information[†]). This magnetic entropy depends relatively weakly on the low temperature cut-off of the fit ($\pm 20\%$ at the extreme ends of the range)[†]. When considering the various possible models for describing the phonon background this magnetic entropy at 60 K is at most 70 % of $R \ln(2S + 1)$ per mole osmium[†]. The same phonon fit was subtracted from data for another sample, yielding a value for magnetic entropy which approaches $S = 5.0 \text{ J mol}^{-1} \text{ K}^{-1}$.

Taken together, the multiple magnetic transitions as a function of temperature, sharp metamagnetic transition with double hysteresis loop, and residual low- T entropy indicate a complex magnetic phase diagram with potential interplay between competing interactions. These features suggest a richer phase space than in the ruthenate analogue. The higher Néel temperature and greater Weiss constant compared to the ruthenate points to stronger antiferromagnetic interactions within $\text{La}_7\text{Os}_3\text{O}_{18}$, likely arising due to the more spatially extended Os d-orbitals. The broad peak we observe in the susceptibility data of $\text{La}_7\text{Os}_3\text{O}_{18}$ at 14 K and associated crossover in the heat capacity data constitute a marked departure from the ruthenate analogue, and taken together sug-



gest the possibility of two interwoven magnetic sublattices, with one having low dimensionality. These surprising findings highlight the unusual phases which can emerge when Os, a 5d metal with strong spin-orbit coupling, is incorporated into complex oxide structures. Our findings motivate further studies on both $\text{La}_7\text{Os}_3\text{O}_{18}$ and $\text{La}_7\text{Ru}_3\text{O}_{18}$, in either case magnetic studies using single crystals would offer insights on the unusual magnetic properties observed. In the case of $\text{La}_7\text{Os}_3\text{O}_{18}$, neutron powder diffraction will be used to shed light on its magnetic phase diagram.

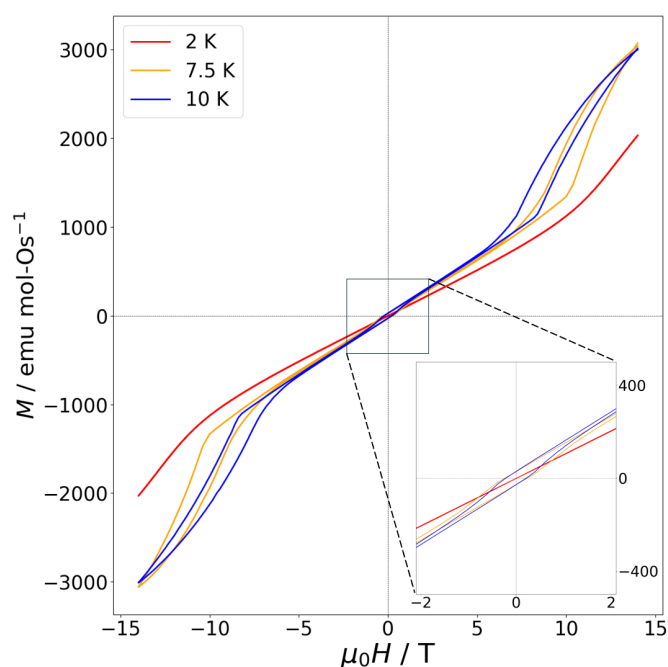


Fig. 3 Measurements of magnetisation versus field for a sample of $\text{La}_7\text{Os}_3\text{O}_{18}$ at 2 K, 7.5 K and 10 K. A distinctive hysteretic field induced magnetic phase transition is observed at 7.5 K and 10 K but is beyond the range of fields measured at 2 K. The inset shows a smaller hysteresis loop in the low-field data at the higher temperature points. A plot of the derivatives is available in the Supplementary Information†.

Conflicts of interest

There are no conflicts to declare.

Acknowledgments

We acknowledge funding from the Engineering and Physical Sciences Research Council via EP/P024564/1 (AWR), EP/T011130/1 (ASG) as well as the International Max Planck Research School for Chemistry and Physics of Quantum Materials (MTP) and for characterisation infrastructure EP/V034138/1, EP/L017008/1, EP/R023751/1, EP/T019298/1 and EP/T031441/1.

Data availability

Data for this article have been included as part of the ESI.

Notes and references

- 1 A. J. Browne, A. Krajewska and A. S. Gibbs, *Journal of Materials Chemistry C*, 2021, **9**, 11640–11654.
- 2 D. I. Khomskii, *Transition Metal Compounds*, Cambridge University Press, Cambridge, 2014.
- 3 Y. Shi, Y. Guo, X. Wang, A. J. Princep, D. Khalyavin, P. Manuel, Y. Michiue, A. Sato, K. Tsuda, S. Yu, M. Arai, Y. Shirako, M. Akaogi, N. Wang, K. Yamaura and A. T. Boothroyd, *Nature Materials*, 2013, **12**, 1024–1027.
- 4 S. A. Mohitkar, J. Nuss, H. A. Höpfe, C. Felser and M. Jansen, *Dalton Transactions*, 2018, **47**, 5968–5976.
- 5 S. Yonezawa, Y. Muraoka and Z. Hiroi, *Journal of the Physical Society of Japan*, 2004, **73**, 1655–1656.
- 6 S. Yonezawa, Y. Muraoka, Y. Matsushita and Z. Hiroi, *Journal of the Physical Society of Japan*, 2004, **73**, 819–821.
- 7 S. Yonezawa, Y. Muraoka, Y. Matsushita and Z. Hiroi, *Journal of Physics Condensed Matter*, 2004, **16**, L9.
- 8 K. Momma and F. Izumi, *Journal of Applied Crystallography*, 2011, **44**, 1272–1276.
- 9 P. Khalifah, Q. Huang, D. M. Ho, H. W. Zandbergen and R. J. Cava, *Journal of Solid State Chemistry*, 2000, **155**, 189–197.
- 10 H. A. Mons, M. S. Schriever and W. Jeitschko, *Journal of Solid State Chemistry*, 1992, **99**, 149–157.
- 11 O. C. Gagné and F. C. Hawthorne, *Acta Crystallographica Section B Structural Science, Crystal Engineering and Materials*, 2015, **71**, 562–578.
- 12 F. C. Goerigk and T. Schleid, in *27th Annual Conference of the German Crystallographic Society, March 25–28, 2019, Leipzig, Germany*, De Gruyter, Berlin, Boston, 2019, p. 93.
- 13 T. Hummel, F. Salk, M. Ströbele, D. Ensling, T. Jüstel and H.-J. Meyer, *European Journal of Inorganic Chemistry*, 2015, **2015**, 977–981.
- 14 Q. Ye, M. Barré, K. Adil, A. Rousseau, E. Suard and F. Goutenoire, *Journal of Solid State Chemistry*, 2022, **313**, 123310.
- 15 J. K. Lee, A. Nande, A. A. Bhat, S. Watanabe, T. K. Rao and V. Singh, *Ceramics International*, 2024, **50**, 17063–17074.
- 16 K. Li and R. V. Deun, *Dyes and Pigments*, 2019, **162**, 990–997.
- 17 K. Li and R. V. Deun, *Materials Chemistry Frontiers*, 2019, **3**, 403–413.
- 18 G. A. Bain and J. F. Berry, *Journal of Chemical Education*, 2008, **85**, 532.
- 19 P. Khalifah, D. A. Huse and R. J. Cava, *Journal of Physics Condensed Matter*, 2003, **15**, 5747.
- 20 A. Vasiliev, O. Volkova, E. Zvereva and M. Markina, *npj Quantum Materials*, 2018, **3**, 18.



Data availability: Data for this article have been included as part of the ESI

[View Article Online](#)
DOI: 10.1039/D5CC06230F

