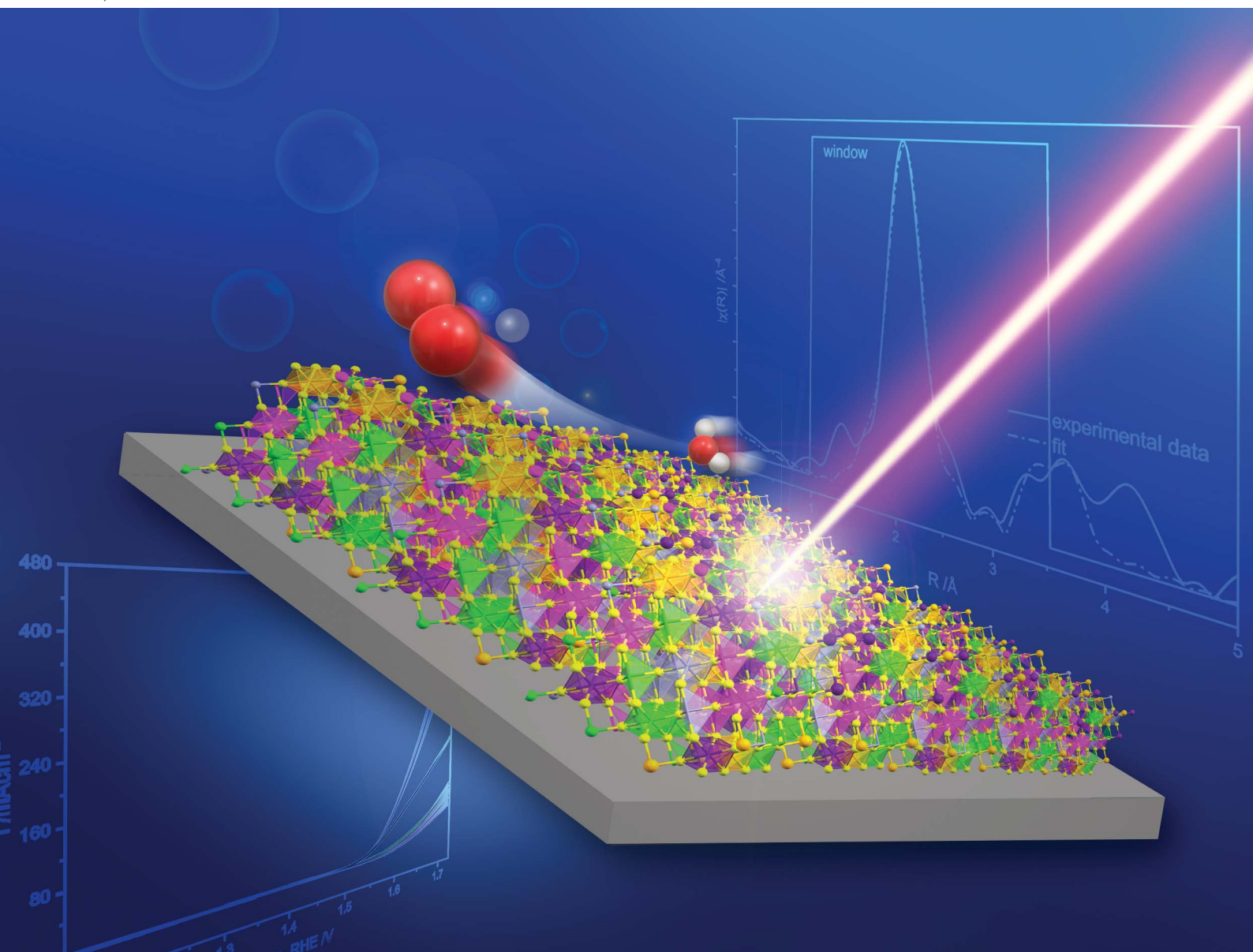


# Journal of Materials Chemistry A

Materials for energy and sustainability

rsc.li/materials-a














ISSN 2050-7488

**PAPER**

Thomas Braun, Franziska Emmerling *et al.*  
Zirconium fluoride-supported high-entropy fluoride: a  
catalyst for enhanced oxygen evolution reaction

Cite this: *J. Mater. Chem. A*, 2025, 13, 20383

# Zirconium fluoride-supported high-entropy fluoride: a catalyst for enhanced oxygen evolution reaction†

Christian Heinekamp, <sup>ab</sup> Arkendu Roy, <sup>ab</sup> Stephanos Karafiludis, <sup>ab</sup> Sourabh Kumar, <sup>a</sup> Ana Guilherme Buzanich, <sup>a</sup> Tomasz M. Stawski, <sup>a</sup> Aistė Miliūtė, <sup>ab</sup> Marcus von der Au, <sup>a</sup> Mike Ahrens, <sup>b</sup> Thomas Braun <sup>\*b</sup> and Franziska Emmerling <sup>\*ab</sup>

Extended hydrogen initiatives promote the urgency of research on water splitting technologies and, therein, oxygen evolution reaction catalysts being developed. A route to access a ZrF<sub>4</sub> supported high-entropy fluoride catalyst using a facile sol-gel route is presented. The high-entropy character of the catalyst was confirmed by scanning transmission electron microscopy and energy dispersive X-ray spectroscopy (STEM-EDX) as well as inductively coupled plasma-mass spectrometry (ICP-MS). Additional investigations on the local structure were performed using extended X-ray absorption fine structure spectroscopy (EXAFS) and pair distribution function (PDF) analysis. The catalyst shows significant potential for oxygen evolution reaction (OER) in alkaline media with a current density of 100 mA cm<sup>-2</sup> at approximately 1.60 V, thus outperforming benchmark materials such as IrO<sub>2</sub>, despite a significant reduction in electrochemical mass loading. A potential mechanism is suggested based on free energy calculation using DFT calculations.

Received 5th December 2024  
Accepted 1st April 2025

DOI: 10.1039/d4ta08664c

rsc.li/materials-a

## Introduction

Climate change is accelerating, prompting a growing need for new energy sources and reduced dependence on fossil fuels.<sup>1,2</sup> In response, many countries have implemented “hydrogen strategies” at the political level, which refers to strategies to implement hydrogen as a new fuel in energy and heat consuming processes.<sup>3</sup> Alongside these efforts, research on water splitting technologies has intensified.<sup>4</sup> Within this field, the development of stable and non-precious metal catalysts has become a major focus due to concerns about cost-efficiency and the limited supply of noble metals.<sup>5–7</sup> This trend coincided with the discovery of high-entropy materials (HEMs) in 2004.<sup>8,9</sup> These complex materials, composed of five or more elements in near-equimolar ratios, have attracted considerable interest in the search for water splitting catalysts.<sup>10</sup> The high-entropy effect offers increased thermodynamic phase stability due to the mixture of elements. Additionally, the unique properties arising from the high-entropy effect in HEMs can enhance their

catalytic activity compared to single-element catalysts. Furthermore, unlike noble metals, which are often scarce and expensive, HEMs can be composed of abundant elements, significantly reducing costs.<sup>11</sup> Given these advantages, the potential of HEMs for various industrial applications is undeniable.<sup>12</sup>

The main reasons for catalytic activity are due to the reduction of the activation energy of adsorption of reagents and stabilisation of intermediates.<sup>13</sup> This becomes particularly relevant when more than one intermediate complicates the reaction mechanism. HEMs directly target these issues by the available high-entropy surface through the modified local structure.<sup>10,14</sup> In the electrochemical water oxidation, the H<sub>2</sub> production at the cathode is limited due to the “sluggish” kinetics at the anode for the oxygen evolution reaction (OER). Thus, improving this reaction side allows for a significant improvement in hydrogen production.<sup>15,16</sup> Evidently, several types of HEM have proven to be of significant contribution to finding a suitable catalyst for OER. For example, Zhao *et al.* and Yu *et al.* have demonstrated the performance of high-entropy alloys supported on conducting MoC and Ni<sub>2</sub>P, respectively.<sup>17,18</sup> Further examples of high-entropy layered hydroxides,<sup>19–21</sup> -oxides,<sup>22–25</sup> -ceramics<sup>26</sup> and -metal organic frameworks<sup>27,28</sup> have been investigated. In particular, Roy *et al.* were able to specifically reduce the amount of active metal sites in their catalyst (≈60% OER active metals), while maintaining a current density of 100 mA cm<sup>-2</sup> at ≈1.64 V<sub>RHE</sub>.<sup>28</sup> However, the field of high-entropy fluorides<sup>29,30</sup> remains relatively unexplored.

<sup>a</sup>Department Materials Chemistry, Federal Institute for Material Research and Testing, Richard-Willstätter-Straße 11, 12489 Berlin, Germany. E-mail: franziska.emmerling@bam.de

<sup>b</sup>Institut für Chemie, Humboldt Universität zu Berlin, Brook-Taylor-Straße 2, 12489 Berlin, Germany. E-mail: thomas.braun@cms.hu-berlin.de

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4ta08664c>



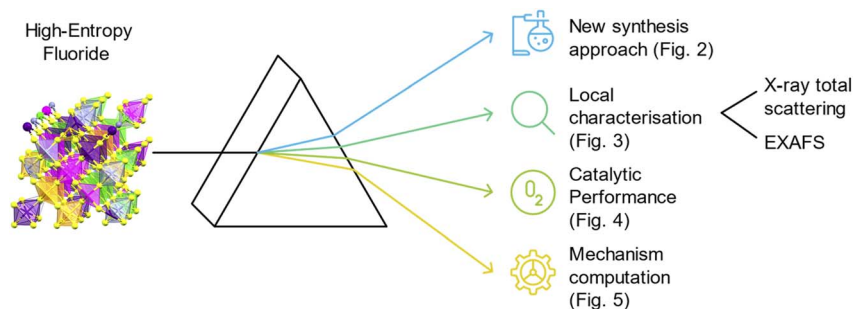


Fig. 1 Schematic description of the presented high-entropy fluoride (Mn: violet, Fe: orange, Co: pink, Ni: green, Zn: grey, F: yellow) and the list of the performed synthesis and analysis in this work. It is also shown where the data is presented.

Herein, we report on a two-step sol-gel route to access a high-entropy fluoride consisting of Mn, Fe, Co, Ni and Zn on a  $ZrF_4$ -support (Fig. 1), which exhibits an overpotential of 317 mV and stability of 50 h at  $10 \text{ mA cm}^{-2}$ . Furthermore, the content of elements involved in the catalysis is minimised to 8 wt% in the synthesised material. Detailed structural analysis, together with theoretical calculation, provided insight into active species and a potential reaction mechanism.

## Methods

The metal acetylacetonates (acac)  $Mn(acac)_2$  (prod. 245 763),  $Fe(acac)_3$  (97%),  $Co(acac)_2$  ( $\geq 99\%$ ), and zinc acetate  $Zn(OAc)_2$  (99,99%) were purchased from Sigma-Aldrich.  $Ni(acac)_2$  (90–95%) was supplied from Fluka.  $Zr(acac)_4$  (98%+) was purchased from ChemPur.  $^iPrOH$  (99.5%, extra pure) was supplied by Carl-Roth, distilled and stored over molecular sieves. Aqueous HF (28.7 M) solution was received from Solvay.

### Preparation of the catalyst

1 g (0.75 eq.) of the  $Zr(acac)_4$  precursor was weighed in under atmospheric conditions and the reaction vessel was purged with Ar for 5 min before the procedure was continued. 100 mL of water free  $^iPrOH$  were then added to the reagent. The mixture was refluxed for 1 h and allowed to cool down. Three equivalents of aq. HF ( $28.7 \text{ mol L}^{-1}$ ) were added, and a milky gel was formed, which was stirred for 15 min. In parallel, the other precursors (35 mg (0.05 eq.)  $Mn(acac)_2$ , 49 mg (0.05 eq.)  $Fe(acac)_3$ , 35 mg (0.05 eq.)  $Co(acac)_2$ , 35 mg (0.05 eq.)  $Ni(acac)_2$  and 25 mg (0.05 eq.)  $Zn(OAc)_2$ ) were added in another vessel and also purged 5 min with Ar. 40 mL of water free  $^iPrOH$  were added and the mixture was refluxed for 10 min to ensure that all reactants were dissolved. The solution was then transferred to the aged Zr gel under inert conditions. The mixture was again refluxed for 30 min to ensure homogeneity. Subsequently, 1.5 eq. aq. HF were added to the mixture. The solvent was removed under a vacuum after the mixture was aged overnight under vigorous stirring. The obtained pale pink product was stored under ambient conditions. Monometallic fluorides on the  $ZrF_4$ -support were synthesised using the same procedure while maintaining the overall Zr to metal ratio at 75% : 25%.

### X-ray diffraction (XRD)

X-ray diffraction patterns were collected using Cu  $K\alpha$  radiation on a Bragg Brentano D8 Advanced diffractometer (Bruker AXS, Germany) equipped with a LYNXEYE XE-T detector. Samples were measured in reflection geometry in a  $2\theta$  range from  $5^\circ$  to  $90^\circ$  with a step size of  $0.02^\circ$  using a spinning setup.

### Transmission electron microscopy (TEM)

High-resolution transmission electron microscopy (HRTEM), high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) and energy dispersive X-ray analysis (EDX) elemental mapping were performed on an FEI Talos F200S scanning/transmission electron microscope (S/TEM) at an acceleration voltage of 120 kV. Dry TEM grid preparation was used. Therefore, TEM grids were carefully swiped across the powder samples. The excess of powder on the grids was removed by tapping lightly.

### TGA/DSC

The TGA and DSC measurements were performed at a TGA/DSC 3+ from Mettler Toledo, Switzerland in Al crucibles. The samples were heated from 25 to  $600^\circ\text{C}$  at a rate of  $10 \text{ K min}^{-1}$ . The samples were then cooled down to  $25^\circ\text{C}$  at the same rate.

### Infrared spectroscopy (IR)

The IR-spectra were recorded on a Bruker Vertex 70 spectrometer equipped with a Golden Gate diamond ATR (attenuated total reflectance) module.

### Extended X-ray absorption fine structure spectroscopy (EXAFS)

EXAFS measurements were performed at the BAMline at BESSY-II.<sup>31</sup> The beam was monochromatized using a double crystal monochromator (DCM) installed at the beamline, with a resolution ( $\Delta E/E$ ) of about  $2 \times 10^{-4}$ . The slits were adjusted to provide a  $4 \text{ mm (H)} \times 1 \text{ mm (V)}$  spot size. The measurements were performed at the Zr K-edge ( $17998 \text{ eV}$ ) in transmission mode. Optimal absorption was achieved by choice of the appropriate sample thickness, leading to an edge jump factor of about 2. This was achieved by diluting the powder samples with boron nitride (BN). Further measurements (at K-edges of Mn –  $6539 \text{ eV}$ , Fe –



7112 eV, Co – 7709 eV, Ni – 8330 eV, Zn – 9659 eV) were performed in fluorescence mode as the Zr content did not allow any X-ray transmission. The excitation energy was varied from –200 eV to –20 eV below the edge in 10 eV steps, from –20 eV below the edge and 200 eV above the edge in 1 eV steps (for Zr K-edge and in 0.25 eV steps for all others), and in the EXAFS region with a constant step in the  $k$ -space of  $0.04 \text{ \AA}^{-1}$  until  $k = 16 \text{ \AA}^{-1}$ . EXAFS data were processed by ATHENA and ARTEMIS.<sup>32</sup> These GUIs programs belong to the main package IFEFFIT (v. 1.2.12).<sup>33</sup> The AutoBK background subtraction procedure was used with the  $R_{\text{bkg}}$  parameter set to  $1.0 \text{ \AA}$  and  $k\omega = 1$ . Afterwards, all spectra were normalized to the far post-edge region, free from absorption features. Regarding the EXAFS region, with ATHENA, one can plot  $\chi(k)$  against  $R(\text{\AA})$ , and the oscillations represent different frequencies, which correspond to the different distances for each coordination shell. Hence, Fourier transforms (FT) are necessary for the analysis process. The FT from the  $k$ -space to the  $R$ -space was performed with a Hanning-type window with a range of  $1.5 \text{ \AA}$  to  $12 \text{ \AA}$  and  $dk = 1$ . By analysing the signal in the frequency domain in ATHENA the window range was selected to exclude the noisy part of the signal.

### Total scattering/pair distribution function analysis (PDF)

High-energy X-ray diffraction (HEXD) measurements were performed at the beamline ID15-A of the ESRF (France). The diffraction patterns were collected at 70 keV, corresponding to a wavelength of  $0.1771203 \text{ \AA}$ . The 2D patterns were measured with a Pilatus3 X CdTe 2M detector in Debye-Scherrer geometry. The powder samples were measured in rotating capillaries using the standard ID15-A setup. The  $q$ -range was calibrated using a  $\text{CeO}_2$  standard. Dark current contributions were corrected automatically by the acquisition software. We obtained the atomic pair distribution functions (PDFs) from the diffraction patterns using the PDFgetX3 software,<sup>34</sup> which was also used for the background and Compton scattering subtractions. Further analysis was performed in Python using a DiffPy library.<sup>35</sup>

### Computational methodologies

First-principles calculations are performed using the projector-augmented wave (PAW) method implemented in the Vienna *Ab Initio* Simulation Package (VASP).<sup>36,37</sup> The Perdew–Burke–Ernzerhof (PBE) generalized gradient approximation (GGA) is used for exchange–correlation energy.<sup>38,39</sup> A plane-wave cutoff energy of 500 eV is applied to truncate the wave functions. Structural relaxations are carried out using the conjugate gradient algorithm, with an energy convergence criterion of  $10^{-6}$  eV for structural optimization and  $10^{-8}$  eV for electronic structure and total energy calculations. The force convergence criterion is set at  $10^{-2} \text{ eV \AA}^{-1}$  for individual atoms. The high-entropy fluoride (HEF) structure is modelled using the special quasi-random structure (SQS) generator code.<sup>40</sup> The HEF structure is based on a rutile-type motif ( $\text{MF}_2$ ), with a unit cell comprising 40 metal (M) atoms and 80 fluorine (F) atoms, considering the (001) surface. A vacuum region exceeding  $15 \text{ \AA}$  is included to prevent interactions between periodic images. The free energy change  $\Delta G$  for an electrochemical reaction is calculated using the following expression:

$$\Delta G = \Delta E_{\text{DFT}} + T\Delta S + \Delta E_{\text{ZPE}} + \Delta G_{\text{U}}$$

Here,  $\Delta E_{\text{DFT}}$  represents the DFT energy change for the reaction (from reactants to products),  $T$  is the temperature (298.15 K),  $\Delta S$  accounts for the entropy contribution,  $\Delta E_{\text{ZPE}}$  is the zero-point energy correction, and  $\Delta G_{\text{U}}$  is the overpotential of the reaction. Phonon calculations are performed using the finite difference method to refine the DFT energies to calculate free energy. These calculations incorporate zero-point energy and entropy corrections at 298.15 K.<sup>28,41–45</sup>

### Electrochemical measurements

Electrodes for electrochemical measurements were prepared by drop-casting a catalyst-ink containing the respective materials containing powders on a carbon paper electrode. Typically, the catalyst ink was prepared with 5 mg of the catalyst and 1 mg of volcanic carbon black in 980  $\mu\text{l}$  water/ethanol (1 : 1) with 20  $\mu\text{l}$  5 wt% Nafion solution, followed by sonication in an ultra-sonic water bath for 1 h. The electrochemical measurements were performed using Hg/HgO reference electrode and Pt counter electrode in an H-cell set-up in 1 mol  $\text{L}^{-1}$  NaOH electrolyte solution. The linear sweep voltammetry (LSV) measurements were performed at a scan rate of  $5 \text{ mV s}^{-1}$ . Ohmic losses in the electrochemical measurements were corrected by subtraction of the ohmic drop from the measured electrolyte resistance, which is determined by high frequency alternating current impedance, where 85%  $iR$  corrected potentials are denoted as  $E$  (V). All measured potentials were converted to values relative to an RHE according to the equation:  $E$  (V) =  $E_{\text{ref}}$  +  $E$  (ref vs. RHE), with  $E$  (ref vs. RHE) the potential difference between the reference electrode and RHE.

The chronopotentiometry was performed in the identical setup at  $10 \text{ mA cm}^{-2}$ .

The electrochemical active surface area (ECSA) was obtained by measuring the double-layer charging current ( $i_c$ ) resulting from scan-rates ( $\nu$ ) dependent CVs at non-faradic potential (0.02–0.08 mV vs. Hg/HgO) and determining the  $C_{\text{dl}}$ .  $C_s$  corresponds to the specific capacitance value of  $12.16 \text{ mF cm}^{-2}$ .<sup>46</sup> The average of the positive and the negative slopes was taken.

$$\text{ECSA} = \frac{C_{\text{dl}}}{C_s}$$

$$i_c = C_{\text{dl}} \times \nu$$

Electrochemical impedance spectroscopy (EIS) measurements were made in the frequency ranges from  $10^5$  to 1 Hz at 1.55 V vs. RHE.

The faradaic efficiency  $\varepsilon$  was determined using a Rotating Ring-Disk electrode setup at a rotational speed of 1600 rpm. The cell was first purged with  $\text{N}_2$  until saturation was achieved. The disk and the Pt-ring current were integrated over 5 min, respectively. Together with the collection efficiency (0.19), the efficiency was calculated using the following equation, where we followed the previous procedure by McCrory *et al.*<sup>47</sup>



$$\varepsilon = \frac{2i_r}{i_d n F}$$

The produced O<sub>2</sub> was calculated by the following equation, where  $Q$  is the transferred charge,  $z$  is the number of electrons needed and  $F$  is the Faraday constant:

$$n = \frac{Q}{F \times z}$$

### ICP-MS analysis

Consumables (50 mL tubes, pipette tips) were cleaned with diluted double sub-boiled nitric acid ( $w = \text{approx. } 3\%$ ) prior to analysis. Type I reagent-grade water (18.2 M $\Omega$  cm, HQ-water) was obtained from a Milli-Q Integral water purification system (Merck-Millipore, Darmstadt, Germany). Analytical reagent-grade nitric acid ( $w = 65\%$ , Merck, Darmstadt, Germany) was purified by double sub-boiling using a sub-boiling distillation system.

Approx. 20 mg of the sample was dissolved in a mixture of 2 mL of double sub-boiled nitric acid ( $w = 65\%$ ) and 200  $\mu\text{L}$  sub-boiled hydrofluoric acid ( $w = 40\%$ , Merck, Darmstadt, Germany). The digest was diluted with HQ-water to a final volume of 50 mL. ICP-MS measurements of the sample were performed in triplicates after the dilution by a factor of 1000 with diluted nitric acid ( $w = 2\%$ ). Multi-elemental analyses (incl. Mg, Cr, Fe, Mn, Co, Ni, Cu, Zn and Zr) of the diluted HEF-samples were performed using an ICP-MS (ICAP-Q, Thermo Scientific, Bremen, Germany). The ICP-MS instrument was optimised in STD-mode in a daily routine using a tuning solution (Thermo Scientific) to obtain maximum signal intensity while maintaining high stability, low oxide and double-charged rates.

General instrumental settings for the multi-elemental measurements are described in ESI Table 1.†

The multi-elemental quantification of the diluted HEF-samples was accomplished by external calibration (4-point calibration ranging for Mg, Cr, Fe, Mn, Co, Ni, Cu and Zn from 5  $\mu\text{g L}^{-1}$  to 50  $\mu\text{g L}^{-1}$  and for Zr from 50  $\mu\text{g L}^{-1}$  to 500  $\mu\text{g L}^{-1}$ ) with standards prepared volumetrically from a multielement stock solution (ICP multielement standard Merck IV, 23 Elements in nitric acid 1000 mg L<sup>-1</sup>) and a zirconium single element standard (Certipur, 1000 mg L<sup>-1</sup>, Merck, Darmstadt, Germany) including 1  $\mu\text{g L}^{-1}$  yttrium (prepared from an Y stock solution 1000 mg L<sup>-1</sup>, Merck, Darmstadt, Germany) as internal normalisation standard. Furthermore, the performance of the multi-elemental analysis was monitored with in-house quality control solution (incl. Mg, Cr, Fe, Mn, Co, Ni, Cu and Zn and Zr) prepared in HNO<sub>3</sub> ( $w = 2\%$ ).

## Results and discussion

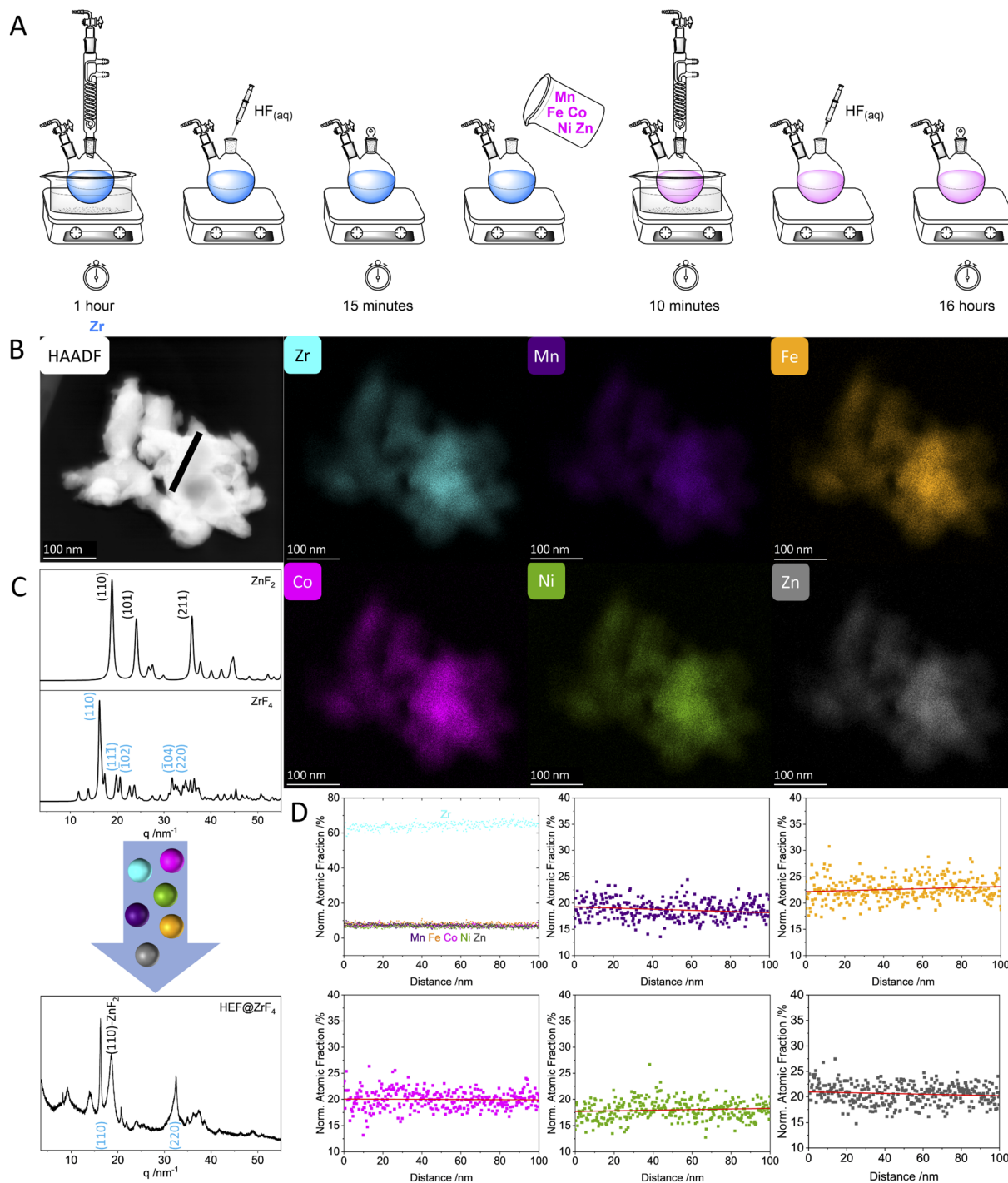
A nanocrystalline high-entropy fluoride (M = Mn, Fe, Co, Ni, Zn) on a zirconium fluoride support was synthesised using an adapted strategy for xerogel synthesis previously described for other metal fluorides.<sup>48,49</sup> In the first step, the support was synthesised by adding three equivalents of an aqueous HF solution to a solution of zirconium acetylacetonate (Zr(acac)<sub>4</sub>) in dry isopropanol forming a mostly amorphous material,

although a small broad reflection is observed in the PXRD that could not be attributed to any known Zr fluoride or oxide phase (see ESI†). In the second step, after gelation a warm solution of the other metal precursors also in dry isopropanol was added to the gel and additional HF in a water solution was added after homogenisation (Fig. 2A). The mixture was homogenised by reheating the solution for 10 min. After subsequent gelation the solvent was removed. In this synthesis, the acac-ligand is protonated by the added HF and the H<sub>2</sub>O in a competitive manner, as it has been described in detail for similar sol-gel routes to obtain the respective metal fluorides.<sup>48,50</sup> The polymeric behaviour of the fluoride allows for bridging between the ZrF<sub>4</sub> support and the HE metal fluoride surface.

### Structural characterisation

The elemental homogeneity of the material was confirmed by elemental STEM mappings (Fig. 2B and D), which show a close to equimolar distribution of Mn, Fe, Co, Ni and Zn on the particles. This is consistent with the ICP MS findings (see ESI†). The PXRD pattern obtained for the prepared HEF displays a two-phase system, where the higher crystalline phase can be assigned to ZrF<sub>4</sub> while the second phase indicates a nanocrystalline entity of the respective rutile-type metal fluorides (MF<sub>2</sub>). The use of a Fe<sup>3+</sup> precursor is likely to introduce additional bias due to the different structure of the corresponding fluoride, resulting in a broader and less defined PXRD pattern.<sup>51</sup> In addition, the low content of elements forming this phase makes it difficult to analyse the structure by PXRD alone. TEM and selected area electron diffraction (SAED, see ESI†) confirmed the interplanar distances observed in the PXRD pattern (Fig. 2C), which also indicates the presence of the (110) planes of ZrF<sub>4</sub> and the rutile MF<sub>2</sub>. Therefore, EXAFS measurements were performed to better understand the local structure. Here, the presence of Fe(III) was confirmed by the absorption edge at 7127 eV (Fig. 3). The small pre-peak at 7115 eV has also been reported for octahedral FeF<sub>3</sub> species.<sup>52</sup> The first coordination sphere appears more distorted compared to the single metal counterpart or the reagent Fe(acac)<sub>3</sub>, as the magnitude of  $\chi(R)$  is broadened and lower in magnitude (see ESI-Fig. 11†). Nevertheless, an octahedral FeF<sub>3</sub> coordination could be confirmed (Fig. 3B bottom left, see ESI Table S8†).<sup>53</sup> For Co and Zn, the expected respective oxidation state +II was also confirmed by the edges at 7719 eV and 9665 eV, respectively. A small contribution of Co(III) could be identified only for Co, as indicated by a second maximum at 7725 eV in the derivative of the absorption spectrum. However, the combined Co entities exhibit an octahedral metal(II)F<sub>2</sub>, similar to rutile-type metal fluorides (Fig. 3, see ESI Table S9†).<sup>54</sup> The present pre-peak at 7709 eV reported for CoF<sub>2</sub> (ref. 55) indicates distorted centrosymmetry, as discussed by de Groot.<sup>56</sup> The EXAFS suggests a more warped environment with a lower coordination number for the Ni sites. Therefore, a model with reduced degeneracy of the fluorine atoms resulted in a corresponding lower amplitude of the fit. The corresponding Ni–F distances were not affected by this aspect. Nevertheless, this suggests a distorted and potentially undercoordinated nickel fluoride environment (see ESI



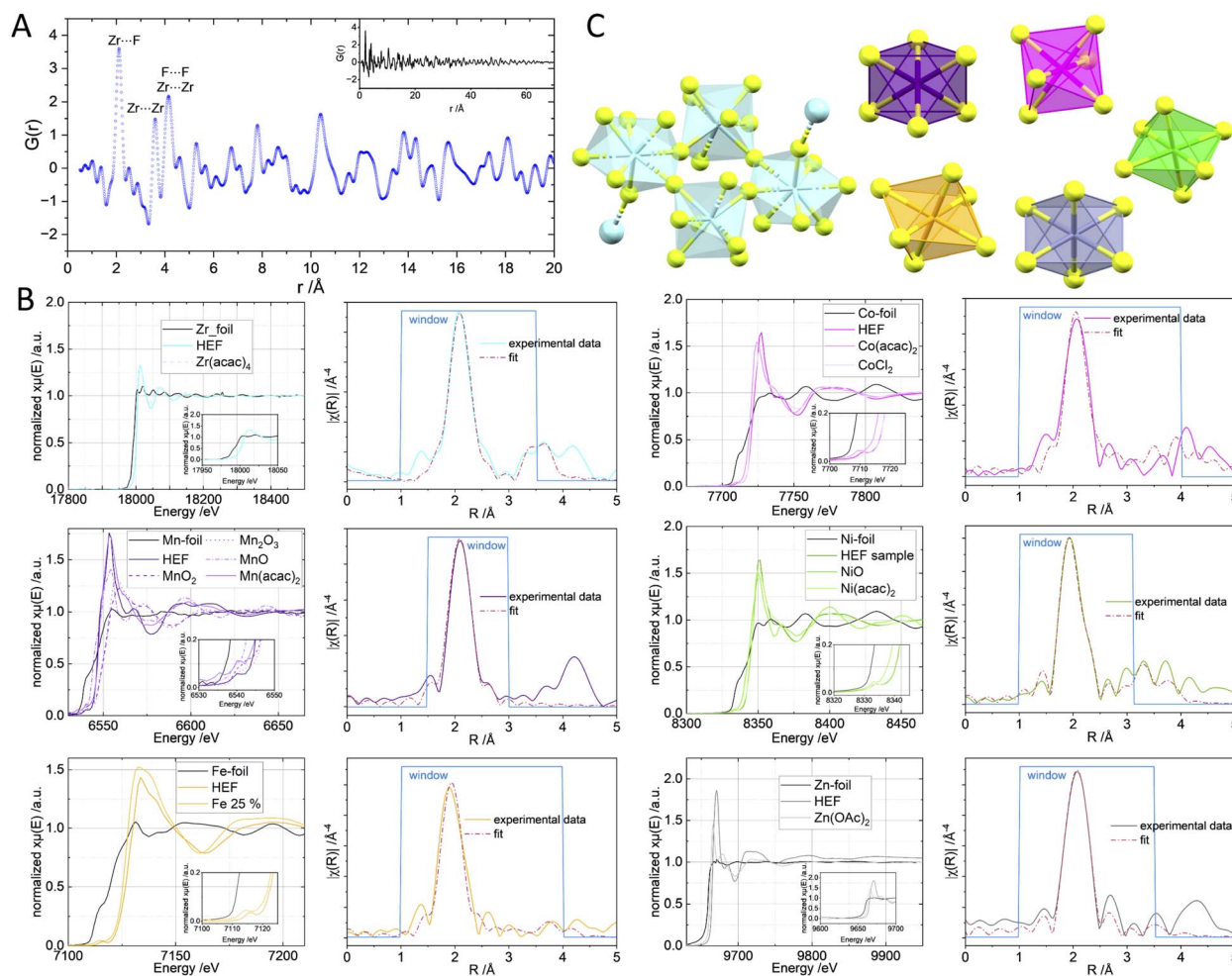


**Fig. 2** Structural and compositional analysis of the HEF. (A) Scheme of the experimental procedure for the synthesis. The reaction scheme can be found in the ESI.† (B) HAADF image of a representative HEF particle and elemental mapping by scanning transmission electron microscopy (STEM) energy dispersive X-ray spectroscopy (EDX) elemental mapping of six different metals (Zr, Mn, Co, Fe, Ni, Zn). (C) Powder X-ray diffraction patterns of the HEF@ZrF<sub>4</sub> and the ZrF<sub>4</sub> and ZnF<sub>2</sub> (rutile type) calculated patterns. (D) Showing the distribution and normalised atomic percentage of the five different metals in the high-entropy entity along the black line drawn on the particle in (B). The Zr representation is normalised to the sum of the HEF contribution and the Zr based support.

Table S5†). Mn sites represent a more complex case with respect to its oxidation state, as Mn has several different stable states available in fluorides<sup>57</sup> as well as in oxides. The XANES data reveal a mixture of two oxidation states. Mn(II) was identified by an edge at 6545 eV, whereas a second Mn species assigned to

6553 eV is present. The comparison to Mn<sub>2</sub>O<sub>3</sub> and MnO<sub>2</sub> suggests an oxidation state of +III (Fig. 3B) as a MnF<sub>4</sub> coordination would shift the absorption edge to higher energies than MnO<sub>2</sub>. Hence, we attribute the value at 6553 eV to an additional Mn(III) fluoride species in the material. A small pre-peak can be





**Fig. 3** (A)  $G(r)$  of the HEF and selected attributed distances of  $\text{ZrF}_4$  in the close environment. The reciprocal data is presented in the ESI.† (B) The normalised K-edge spectra of the six metals including appropriate reference samples. The inset shows a zoom on the respective edge step or the present pre-peak feature. The spectra are also shown in the ESI.† The magnitude plots of the  $\chi(R)$  and the fits of the respective metal(II) fluoride are shown next to the respective K-edge spectrum. In case of Fe the Fe(III) was taken as fitting model. The Zr data was fitted against a  $\text{ZrF}_4$  model. The obtained R values are listed in ESI Table 2.† (C) Structural models of the respective confirmed structural motifs in the  $\text{ZrF}_4$  and the HEF entities.

identified at 6540 eV, similar to what was observed for Mn(II) and Mn(III) in minerals.<sup>58</sup> Despite the different oxidation states present in the sample, the octahedral coordination sphere of  $\text{MnF}_2$  is confirmed by the model used in the EXAFS fit. The radial distribution of the elements did not reveal additional information about further coordination distances around the respective absorbing elements that could directly be attributed to another specific element (Fig. 3B). This observation is expected in an HE material where the second coordination sphere metal distance is broadened due to the random composition. Fitting of the Zr K-edge data revealed the expected eightfold fluorine coordination of the Zr.<sup>59</sup>

In order to generate a more than atomistic model using experimental data, total scattering experiments were performed. Attempts to fit a combined model of  $\text{ZrF}_4$  and the HE-phase using the total scattering data were unsuccessful because the material's small crystallite size and amorphous fraction made establishing a suitable model difficult. Therefore, the  $G(r)$

was investigated qualitatively by comparing the dominant distances to the dominant  $\text{ZrF}_4$  bulk phase (Fig. 3A). Here, the  $\text{Zr}\cdots\text{F}$  distance could be attributed similarly to the EXAFS fit at approx. 2.1  $\text{\AA}$ . The next closer distances at 3.56 and 4.16  $\text{\AA}$  can be attributed equally to  $\text{Zr}\cdots\text{Zr}$  and  $\text{F}\cdots\text{F}$  distances, whereas the partial contribution cannot be determined. Furthermore, the PDF data revealed that the highest contribution of the PDF can be found at about 70  $\text{\AA}$  in accordance with the TEM data (see ESI†), suggesting only very small single crystallites of  $\text{ZrF}_4$  in the particles.

As mentioned above, the characterisation of the high entropy material proved to be difficult due to its complex composition. Therefore, the single metal derivatives were also synthesised. Examination of the XRD data obtained (see ESI†) revealed some differences. While the single metal Fe-containing sample shows the previously reported structure of a  $\text{ZrF}_{4-x}(\text{OH},\text{F})_x$  Xerogel<sup>49</sup> (see ESI†), the single metal Co, Mn and Zn-containing samples are similar to each other except that Co and Mn also exhibit



a  $\text{ZrF}_4 \cdot \text{H}_2\text{O}$  phase. This can be due to the use of aq. HF solution in the synthesis, although such difference is rather unexpected due to the identical treatment during the synthetic procedure.

### Electrochemical OER performance

The obtained high-entropy material was evaluated in a three-electrode cell containing a  $1 \text{ mol L}^{-1}$  NaOH solution at a low scan rate of  $5 \text{ mV s}^{-1}$ . The electrodes were prepared by depositing a homogeneous ink of the materials on an electrochemically inert carbon paper. This allowed the intrinsic activity of the catalysts to be studied instead of using conductive materials such as Ni foams. The HEF on the  $\text{ZrF}_4$  support exhibits higher activity ( $1.60 \text{ V}_{\text{RHE}}$  @  $100 \text{ mA}$ ) than the typical benchmark material  $\text{IrO}_2$  ( $1.67 \text{ V}_{\text{RHE}}$  @  $100 \text{ mA}$ ). The electrochemical surface area (ECSA) was calculated to be  $0.15 \text{ cm}^2$  after estimating the double layer capacitance from the slope in the plot of current density versus the scan rate (Fig. 4E and F). By normalising the geometrical current density by the ECSA one would therefore obtain a specific activity of  $66 \text{ mA cm}^{-2}$ , emphasising the significant activity of the material. The electrochemical impedance spectroscopy (EIS; Nyquist plot) of the HEF material is presented in Fig. 4D and was obtained at  $1.3$  and  $1.55 \text{ V}$  vs. RHE. The data was fitted with a pseudo Randles circuit, resulting in a charge transfer resistance of  $4.784 \Omega$ . The comparison to recent developments in the field of high-entropy catalysts elaborates on the performance of the  $\text{ZrF}_4$  supported material. For instance, aluminium based high-entropy oxides  $\text{Al}_{1-x}(\text{FeCuNiCo})_x\text{O}$  ( $350 \text{ mV}$  @  $10 \text{ mA cm}^{-2}$ ),<sup>60</sup> high-entropy perovskite fluorides ( $315 \text{ mV}$  @  $10 \text{ mA cm}^{-2}$ )<sup>61</sup> as

well as relatively newly reported high entropy molybdates ( $318 \text{ mV}$  @  $10 \text{ mA cm}^{-2}$ )<sup>62</sup> are outperformed. The previously mentioned supported high-entropy alloy supported on MoC shows higher activity ( $232 \text{ mV}$  @  $10 \text{ mA cm}^{-2}$ )<sup>18</sup> compared to our catalyst. It should be noted that their support is conductive, which affects the current density, and that we significantly reduce the number of catalytically active metal sites in our material while maintaining competitive activity. This is revealed by the calculation of the mass activity of the catalyst to be  $80 \text{ mA mg}^{-1}$  at  $1.6 \text{ V}$ , outperforming the  $\text{IrO}_2$  benchmark at only  $56 \text{ mA mg}^{-1}$ . For the control study respective to the enhancement through the compositional complexity, single metal derivatives were investigated. The Mn, Fe, Zn single metal derivatives of the support material (Fig. 4A, see ESI†) exhibit poorer performance. In particular, only the Co and Ni containing single metal derivatives show any significant activity in the investigated OER (Fig. 4A). An equal physical mixture of all single metallic  $\text{ZrF}_4$  supported materials was also tested as an OER catalyst (see ESI-Fig. 2†). The mixture did not exceed the Co and Ni catalysts, confirming the superior nature of the HEF. This is likely due to the so-called “cocktail effect”, which describes the random distribution of the active elements nurturing the activity of the overall material.<sup>63,64</sup> To test whether the complex composition of the catalyst could be simplified to the indicated main catalytic sites originating from Ni and Co a bimetallic fluoride was synthesised. The combination of Ni and Co in one material did not improve the catalytic performance compared to the single metal derivatives (Fig. 4A). However, in subsequent tests the stability of the HE-catalyst was also demonstrated for 50 h (Fig. 4C).

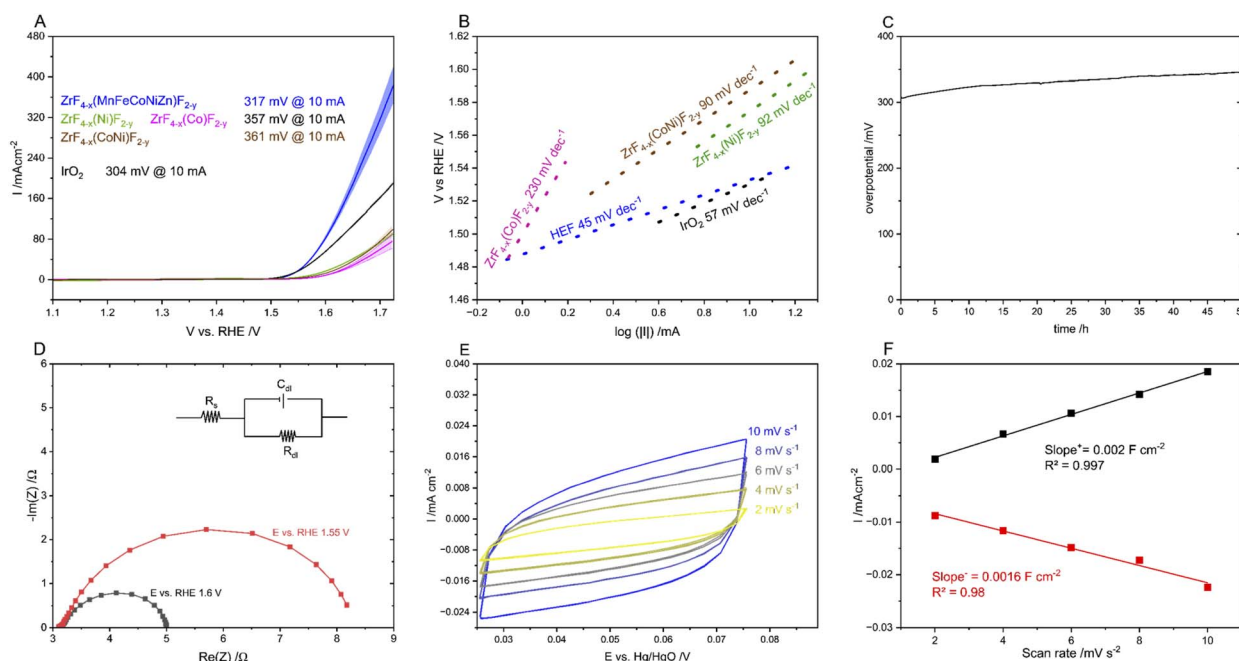


Fig. 4 (A) Linear sweep voltammetry (LSV) of the HEF (blue), the Ni (green) and Co (pink) single metallic derivatives, the Ni–Co bimetallic compound (brown) and an  $\text{IrO}_2$  (grey) reference measurement on conductive carbon paper. (B) Tafel slope plot of the in A presented samples in their respective colours. (C) Chronopotentiometry stability test of the HEF for 50 h. (D) Electrochemical impedance spectroscopy (EIS) Nyquist plot of the HEF material for OER by fixing a constant potential (above the overpotential) at  $1.55 \text{ V}$  (red) and  $1.6 \text{ V}$  (grey) vs. RHE. The corresponding circuit is drawn for representation. (E) Cyclic voltammetry (CV) results in the non-faradaic potential region at scan rate varying from  $2 \text{ mV s}^{-1}$  to  $10 \text{ mV s}^{-1}$ . (F) Current vs. scan rate plot for obtaining the double capacitance.



Another parameter that is not consistently reported is the faradaic efficiency, which indicates how effectively the electrons are being used to produce  $O_2$ . Our system has an efficiency of 0.98, which is remarkable. The corresponding data is shown in the ESI.† Nevertheless, the calculated  $O_2$  produced is about 0.057 mmol. The discrepancy with the theoretical value of 0.080 mmol of  $O_2$  produced can be explained by the saturation of  $O_2$  in the electrolyte, which reduces the oxygen reaching the ring electrode and, thus, the detected  $O_2$ . In addition, other diffusion effects may affect the detected oxygen by reducing the ring electrode current, although the collection efficiency should compensate for these effects. Although the catalyst shows the stability shown, a small amount of Zr, Zn and Co could be detected in the electrolyte. The other elements could not be determined, but were also detected. This is most likely due to

the decomposition of the carbon paper electrode material (see ESI†).

To understand the role of the high-entropy matrix on electrocatalysis, we constructed a free energy diagram to study the OER energetics of the HEF sample, considering possible adsorbate evolution mechanisms. Similarly, to investigate the kinetics of OER, we calculated a Tafel slope of  $\approx 45 \text{ mV dec}^{-1}$  for the HEF matrix (Fig. 4B). Based on previous observations of monometallic  $MF_2$  samples, we considered Co and Ni to be catalytically more active for OER. Other metals such as Mn, Fe, and Zn modify the electronic environment of Co and Ni and stabilize the  $MF_2$  matrix, inducing synergistic effects that ultimately enhance catalyst performance. In the HEF matrix, the stability of the  $M^{2+}/M^{3+}$  redox pair is dynamic, and the  $M^{3+}$  oxidation state is easily accessible to Co in a highly alkaline

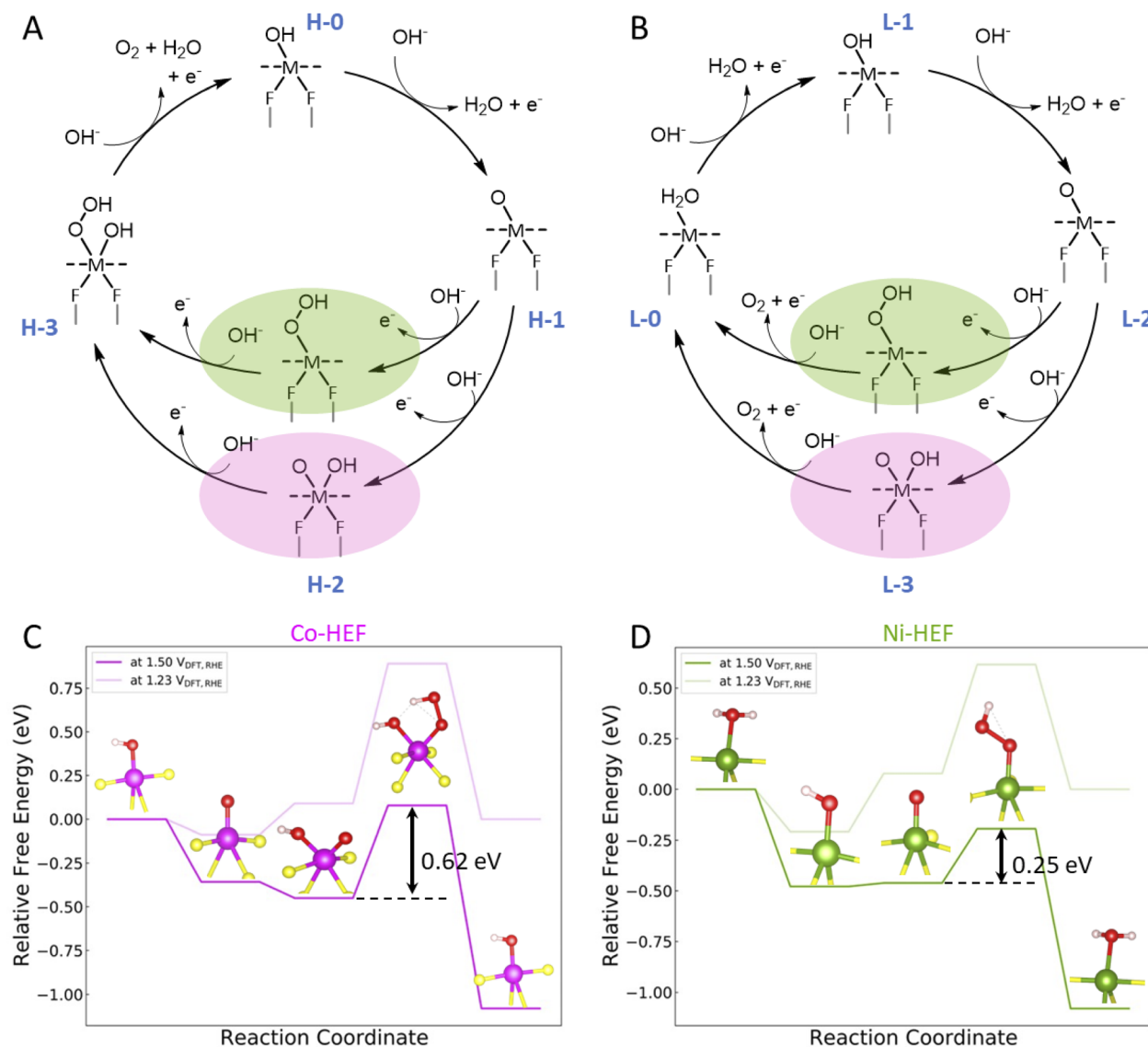


Fig. 5 Reaction-mechanism of OER on HEF-surface: (A and B) proposed adsorbate evolution mechanism for Co (pink) and Ni (green) involving the formation of  $*OOH$  species as a rate-determining step via  $*O-O$  coupling from  $*Ni-O$  and  $*Co-(O)(OH)$  species. "H" and "L" represent suggested oxidation states for metal ions that could be accessible in higher and lower potential regimes ( $V_{RHE}$ ) to derive the kinetics of reactions. (C and D) Free energy diagrams for possible OER active metals (Co and Ni) under consideration, at thermodynamic potential ( $1.23 V_{RHE}$ ) and the potential considered for the calculation of experimental Tafel slope ( $\approx 1.50 V_{RHE}$ ).



solution compared to Ni. Fig. 5C and D illustrate the free energy diagrams based on DFT calculations, comparing the energetics of OER between Co and Ni. The deprotonation of  $\text{Ni}^{2+}(\text{H}_2\text{O})$  to  $\text{Ni}^{3+}(\text{OH})$  species occurs at approximately  $\approx 1.21 V_{\text{DFT,RHE}}$ , which is about  $0.45 V_{\text{DFT,RHE}}$  higher than the  $\text{Co}^{2+}(\text{H}_2\text{O})/\text{Co}^{3+}(\text{OH})$  pair. This suggests further deprotonation of hydroxide species is easily feasible, which may lead to different catalytic sites on the HEF surface with increasing electrode potentials. Our investigations support that at a higher potential, reaching a catalytic cycle between the formal  $\text{Ni}^{3+}/\text{Ni}^{4+}$  pair is equally feasible. Here, we have constructed free energy diagrams specifically for the formal  $\text{Ni}^{2+,3+}/\text{Ni}^{4+}$  and  $\text{Co}^{3+}/\text{Co}^{4+}$  pairs.

Examining the free energy diagram for Co and Ni at  $1.23 V_{\text{DFT,RHE}}$  reveals two uphill reactions. Our *ab initio* simulations confirm that the HEF surface is predominantly covered with  $\text{*OH}$  or  $\text{*O}$  species at around  $1.50 V_{\text{DFT,RHE}}$ . According to the conventional adsorbate evolution mechanism,  $\text{*O-O}$  coupling (peroxide formation) is the most crucial step in OER energetics. For Ni, the reaction proceeds *via* forming a peroxo species from  $\text{*Ni-O}$  to  $\text{*Ni-OOH}$ . In contrast, the formation of oxo/hydroxyl species from  $\text{*Co-O}$  to  $\text{*Co-(O)(OH)}$  is more favored for Co. According to the free energy diagram, the H-3 (Fig. 5) formation is the rate-determining step (RDS), leading to the formation of  $\text{*M-(OOH)(*/OH)}$  species. The same RDS is observed for Ni and Co on the HEF surface.

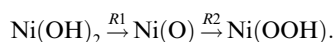
To correlate the observed kinetics—specifically the Tafel slope—with the proposed OER mechanism at the experimental potential of  $\approx 1.50 V_{\text{DFT,RHE}}$ , we consider several assumptions: prior to the RDS, all reactions are in quasi-equilibrium; all OER active species participate in driving the reactions; and after the RDS, reactions only modify the current/overpotential curve.

The Tafel slope ( $b$ )<sup>65</sup> is related to the transfer coefficient ( $\alpha$ ) by the expression

$$b = \frac{2.303 RT}{\alpha F}, \text{ where } (\alpha = n_p + 0.5 \times n_r).$$

$\alpha$  depends on the number of electrons involved in electrochemical reactions prior to the RDS ( $n_p$ ) and at the RDS ( $n_r$ ) for multistep reactions. Assigning a value of  $\alpha = 1.5$  correlates with the observed Tafel slope of  $\approx 45 \text{ mV dec}^{-1}$  around the onset potential at  $\approx 1.50 V_{\text{RHE}}$ .

For Co, considering the formation of  $\text{*Co-(OH)(OOH)}$  as the RDS from  $\text{*Co-(O)(OH)}$  requires only one electron transfer at the RDS ( $n_r = 1$ ). Assuming that all the reactions proceed *via* concerted electron transfer processes, we calculated a Tafel slope of approximately  $\approx 120 \text{ mV dec}^{-1}$ . Similarly, for Ni, considering the formation of  $\text{*Ni-(OOH)}$  requires two concerted electron transfer reactions:



Assigning  $n_p = 1$  and  $n_r = 1$  ( $\alpha = 1.5$ ) results in a calculated Tafel slope of approximately  $\approx 40 \text{ mV dec}^{-1}$ . This emphasizes that the initial kinetics of OER could be triggered by the  $\text{*Ni}$  species at the HEF surface. However, the kinetics of OER are mainly affected by the local environment of the metal, which is

modified by the formation of the high-entropy matrix. This synergistic behavior may lead to more complex intermediates and kinetics. Integrating data from *operando* spectroscopy with *ab initio* calculations in future work has the potential to provide deeper mechanistic insights into HEFs, which could lead to the advancement of OER catalysis.

## Conclusions

In conclusion, this study introduces a novel zirconium fluoride supported high-entropy fluoride (HEF) synthesized *via* a simple sol-gel method. The combination of detailed characterization and *ab initio* calculations provides crucial insights into the material's structure, composition, and catalytic mechanisms. Based on the combined investigations from XRD, ICP-MS and EDX-STEM studies the homogeneity of the HEF sample is shown. The identification of the local structure of the HEF compounds was probed by EXAFS and PDF analysis.

Despite its low content of electrochemically active elements, the HEF demonstrates exceptional catalytic performance for the oxygen evolution reaction ( $100 \text{ mA cm}^{-2}$  at  $1.60 V_{\text{RHE}}$ ), outperforming the benchmark  $\text{IrO}_2$  catalyst in terms of overpotential and current density ( $100 \text{ mA cm}^{-2}$  at  $1.67 V_{\text{RHE}}$ ).

*Ab initio* simulations reveal various potential intermediates involved in OER catalysis, highlighting Ni and Co sites as the OER active species. This comprehensive approach, integrating experimental and theoretical methods, provides the necessary knowledge to optimize these materials for future applications, paving the way for efficient, durable, and cost-effective alternatives to noble metal-based catalysts in electrochemical technologies specifically  $\text{H}_2$  production.

## Data availability

The data supporting this article have been included as part of the ESI.†

## Author contributions

CH performed synthesis, characterisation and wrote the original draft. CH and AR conducted electrocatalysis experiments. CH, SK and AGB conducted the XAS measurements and jointly analysed the obtained data. SKu conducted the theoretical calculations. SK, AM and TMS performed the PDF experiment. TMS and CH performed the analysis. MvA performed the ICP-MS analysis. MA, TB and FE supervised the work. All authors contributed to the manuscript and edited it.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

We thank Ian Pedraza Ramirez for the synthesis of the bimetallic Ni-Co-derivative of the catalyst. Leandro Augusto Faustino is appreciated for scientific discussions and help in the



determination of produced O<sub>2</sub>. We thank Jörg Radnik and Emil Fuhry for fruitful scientific discussions. The authors thank Nicole Langhammer for additional ICP experiments on the electrolyte. XAS experiments were performed at the BAMline at the BESSY-II storage ring (Helmholtz Centre Berlin). We thank the Helmholtz-Zentrum Berlin für Materialien und Energie for the allocation of synchrotron radiation beamtime. The X-ray total scattering experiment was performed at ESRF under the proposal (Ref. No. 95886). We acknowledge financial support from the CRC 1349 “Fluorine Specific Interactions” funded by the German Research Foundation (project number 387284271) and from the School of Analytical Sciences Adlershof (SALSA).

## Notes and references

- Intergovernmental Panel On Climate Change (Ipc), *Climate Change 2021 – The Physical Science Basis: Working Group I Contribution to the Sixth Assessment Report of the Intergovernmental Panel on Climate Change*, Cambridge University Press, 1st edn, 2023.
- Intergovernmental Panel On Climate Change (Ipc), *Climate Change 2022 – Impacts, Adaptation and Vulnerability: Working Group II Contribution to the Sixth Assessment Report of the Intergovernmental Panel on Climate Change*, Cambridge University Press, 1st edn, 2023.
- Global Hydrogen Review 2023*, IEA, Paris, 2023.
- L. Sun, Q. Luo, Z. Dai and F. Ma, *Coord. Chem. Rev.*, 2021, **444**, 214049.
- S. M. Jowitt, G. M. Mudd and J. F. H. Thompson, *Commun. Earth Environ.*, 2020, **1**, 13.
- A. Zupanc, J. Install, M. Jereb and T. Repo, *Angew. Chem., Int. Ed.*, 2023, **62**, e202214453.
- J. Newbold, *J. Cleaner Prod.*, 2006, **14**, 248–261.
- B. Cantor, I. T. H. Chang, P. Knight and A. J. B. Vincent, *Mater. Sci. Eng., A*, 2004, **375–377**, 213–218.
- J. W. Yeh, S. K. Chen, S. J. Lin, J. Y. Gan, T. S. Chin, T. T. Shun, C. H. Tsau and S. Y. Chang, *Adv. Eng. Mater.*, 2004, **6**, 299–303.
- Y. Sun and S. Dai, *Sci. Adv.*, 2021, **7**, eabg1600.
- H. Kamaruddin, Z. Jianghong, L. Yu, W. Yuefan and H. Yizhong, *J. Mater. Chem. A*, 2024, **12**, 9933–9961.
- A. Amiri and R. Shahbazian-Yassar, *J. Mater. Chem. A*, 2021, **9**, 782–823.
- E. Roduner, *Chem. Soc. Rev.*, 2014, **43**, 8226–8239.
- J. Åqvist, M. Kazemi, G. V. Isaksen and B. O. Brandsdal, *Acc. Chem. Res.*, 2017, **50**, 199–207.
- J. Song, C. Wei, Z.-F. Huang, C. Liu, L. Zeng, X. Wang and Z. J. Xu, *Chem. Soc. Rev.*, 2020, **49**, 2196–2214.
- S. Z. Oener, A. Bergmann and B. R. Cuenya, *Nat. Synth.*, 2023, **2**, 817–827.
- H. Yu, W. Chen, Z. Liu and C. Du, *Colloids Surf., A*, 2024, **686**, 133354.
- J. Zhao, C. Zhang, Z. Zhang, Q. Qi, Y. Zhang and J. Hu, *Appl. Catal., A*, 2024, **681**, 119780.
- Y. Ding, Z. Wang, Z. Liang, X. Sun, Z. Sun, Y. Zhao, J. Liu, C. Wang, Z. Zeng, L. Fu, M. Zeng and L. Tang, *Adv. Mater.*, 2023, 2302860.
- T. X. Nguyen, C.-C. Tsai, V. T. Nguyen, Y.-J. Huang, Y.-H. Su, S.-Y. Li, R.-K. Xie, Y.-J. Lin, J.-F. Lee and J.-M. Ting, *Chem. Eng. J.*, 2023, **466**, 143352.
- H. Wu, J. Zhang, Q. Lu, Y. Li, R. Jiang, Y. Liu, X. Zheng, N. Zhao, J. Li, Y. Deng and W. Hu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 38423–38432.
- Y. Lin, W. Xu, Z. Gao, Y. Liang, H. Jiang, Z. Li, S. Wu, Z. Cui, H. Sun, H. Zhang and S. Zhu, *Chem. Eng. J.*, 2024, **489**, 151233.
- B. L. Božiček, J. Hreščak, M. Kušter, J. Kovač, I. Naglič, B. Markoli, B. Š. Batič, M. Šala, S. Drev, Ž. Marinko, M. Čeh and B. A. Marinho, *J. Mater. Sci.*, 2024, **59**, 9189–9207.
- S. Praveen Kumar, P. C. Sharafudeen and P. Elumalai, *Int. J. Hydrogen Energy*, 2023, **48**, 38156–38171.
- Y. Gu, A. Bao, X. Wang, Y. Chen, L. Dong, X. Liu, H. Pan, Y. Li and X. Qi, *Nanoscale*, 2022, **14**, 515–524.
- A. Bao, J. Wu, Y. Zhang, X. Wang, X. Zhang, Y. Gu and X. Qi, *J. Mater. Res. Technol.*, 2023, **23**, 1312–1320.
- S. Xu, M. Li, H. Wang, Y. Sun, W. Liu, J. Duan and S. Chen, *J. Phys. Chem. C*, 2022, **126**, 14094–14102.
- A. Roy, S. Kumar, A. Guilherme Buzanich, C. Prinz, E. Götz, A. Retzmann, T. Hickel, B. Bhattacharya and F. Emmerling, *Adv. Mater.*, 2024, 2408114.
- P. A. Sukkurji, Y. Cui, S. Lee, K. Wang, R. Azmi, A. Sarkar, S. Indris, S. S. Bhattacharya, R. Kruk, H. Hahn, Q. Wang, M. Botros and B. Breitung, *J. Mater. Chem. A*, 2021, **9**, 8998–9009.
- P. Yang, Y. An, C. Feng, Y. Liu, S. Liu, L. Gao, Y. Zhou, X. Li, P. Li and F. Zeng, *Int. J. Hydrogen Energy*, 2024, **51**, 1218–1228.
- A. Guilherme Buzanich, M. Radtke, K. V. Yusenko, T. M. Stawski, A. Kulow, C. T. Cakir, B. Röder, C. Naese, R. Britzke, M. Sintschuk and F. Emmerling, *J. Chem. Phys.*, 2023, **158**, 244202.
- B. Ravel and M. Newville, *J. Synchrotron Radiat.*, 2005, **12**, 537–541.
- M. Newville, *J. Synchrotron Radiat.*, 2001, **8**, 322–324.
- P. Juhás, T. Davis, C. L. Farrow and S. J. L. Billinge, *J. Appl. Crystallogr.*, 2013, **46**, 560–566.
- P. Juhás, C. L. Farrow, X. Yang, K. R. Knox and S. J. L. Billinge, *Acta Crystallogr., Sect. A: Found. Adv.*, 2015, **71**, 562–568.
- G. Kresse and J. Hafner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **47**, 558.
- G. Kresse and J. Hafner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **49**, 14251.
- J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1997, **78**, 1396.
- D. Gehringer, M. Friák and D. Holec, *Comput. Phys. Commun.*, 2023, **286**, 108664.
- J. K. Nørskov, F. Studt, F. Abild-Pedersen and T. Bligaard, *Fundamental Concepts in Heterogeneous Catalysis*, Wiley, 1st edn, 2014.
- I. C. Man, H. Su, F. Calle-Vallejo, H. A. Hansen, J. I. Martínez, N. G. Inoglu, J. Kitchin, T. F. Jaramillo, J. K. Nørskov and J. Rossmeisl, *ChemCatChem*, 2011, **3**, 1159–1165.



- 43 *Handbook of Materials Modeling: Applications: Current and Emerging Materials*, ed. W. Andreoni and S. Yip, Springer International Publishing, Cham, 2020.
- 44 J. Rossmeisl, Z.-W. Qu, H. Zhu, G.-J. Kroes and J. K. Nørskov, *J. Electroanal. Chem.*, 2007, **607**, 83–89.
- 45 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jónsson, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.
- 46 R. Yadav, P. H. Maheshwari, P. Dubey and J. S. Tawale, *Diamond Relat. Mater.*, 2024, **149**, 111589.
- 47 C. C. L. McCrory, S. Jung, J. C. Peters and T. F. Jaramillo, *J. Am. Chem. Soc.*, 2013, **135**, 16977–16987.
- 48 C. P. Marshall, G. Scholz, T. Braun and E. Kemnitz, *Dalton Trans.*, 2019, **48**, 6834–6845.
- 49 C. Heinekamp, S. Kneiske, A. Guilherme Buzanich, M. Ahrens, T. Braun and F. Emmerling, *Catal. Sci. Technol.*, 2024, **14**, 673–680.
- 50 E. Kemnitz, U. Gross, S. Rudiger and C. S. Shekar, *Angew. Chem., Int. Ed.*, 2003, **42**, 4251–4254.
- 51 J.-E. Jørgensen and R. I. Smith, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2006, **62**, 987–992.
- 52 A. Kuzmin and P. Parent, *J. Phys.: Condens. Matter*, 1994, **6**, 4395–4404.
- 53 M. Leblanc, J. Pannetier, G. Férey and R. de Pape, *Rev. Chim. Miner.*, 1985, DOI: [10.1002/chin.198536015](https://doi.org/10.1002/chin.198536015).
- 54 W. H. Baur, *Acta Crystallogr.*, 1958, **11**, 488–490.
- 55 J. P. Krogman, J. R. Gallagher, G. Zhang, A. S. Hock, J. T. Miller and C. M. Thomas, *Dalton Trans.*, 2014, **43**, 13852.
- 56 F. De Groot, *Chem. Rev.*, 2001, **101**, 1779–1808.
- 57 T. C. Ehlert and M. Hsia, *J. Fluorine Chem.*, 1972, **2**, 33–51.
- 58 E. Chalmin, F. Farges and G. E. Brown, *Contrib. Mineral. Petrol.*, 2009, **157**, 111–126.
- 59 J.-P. Laval, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2014, **70**, 742–748.
- 60 M. Asim, A. Hussain, S. Kanwal, A. Ahmad, Y. Aykut, A. Bayrakçeken and N. K. Janjua, *Mater. Adv.*, 2024, **5**(21), 8490–8509.
- 61 T. Wang, H. Chen, Z. Yang, J. Liang and S. Dai, *J. Am. Chem. Soc.*, 2020, **142**, 4550–4554.
- 62 H. K. Beere, P. Kulkarni, U. N. Maiti, R. G. Balakrishna, P. Mukherjee, H. Y. Jung, K. Samanta and D. Ghosh, *Sustainable Energy Fuels*, 2023, **7**, 4303–4316.
- 63 X. Zhao, H. Cheng, X. Chen, Q. Zhang, C. Li, J. Xie, N. Marinkovic, L. Ma, J.-C. Zheng and K. Sasaki, *J. Am. Chem. Soc.*, 2024, **146**, 3010–3022.
- 64 D. B. Miracle and O. N. Senkov, *Acta Mater.*, 2017, **122**, 448–511.
- 65 J. O. Bockris and A. K. N. Reddy, *Modern Electrochemistry: an Introduction to an Interdisciplinary Area Volume 1*, Springer US, Boston, MA, 1970.

