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Constructing a Visible-Light-Excited Z-scheme Heterojunction by **Engineering the Directional N-C/Cu Insertion layer: Overcoming** the Work Function Mismatches

Hao Gao, ^{†a} Xiaoxiao He, ^{†b} Jinbu Li, ^{†a} Qiang Zhu, ^a Chengyu Qin, ^a Liming Sun, ^{*a} Shuting Zhi, ^a Lei Yang, ^c Wenwen Zhan,*a Jianwei Zhao,c and Xiguang Han*a

The construction of S-scheme heterojunctions is constrained by stringent work function (Φ) matching between oxidation and reduction photocatalysts, which limits material selection. Here, we present an innovative interfacial engineering strategy to overcome Φ-mismatched barriers by introducing a nitrogen-doped carbon (N-C) mediator and Cu nanoparticles at the WO₃/Cu₂O interface. Through a "post-deposition and pyrolysis" approach, we fabricated a tightly integrated Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction, where the N-C layer and metallic Cu synergistically redirect photogenerated carrier recombination, preserving the high redox potentials of WO₃ (VB: +2.62 V) and Cu₂O (CB: −1.41 V). Femtosecond transient absorption spectroscopy and electron paramagnetic resonance data revealed that interfacial electrons from WO₃ transferred to N-C and recombined with holes originated from Cu₂O on Cu via the directional N-C/Cu insertion layer. The optimized heterojunction exhibits exceptional photocatalytic performance under blue light (450 nm), achieving a 99% yield in homo-coupling of terminal alkyne to 1,3-conjugated diynes and a hydrogen evolution rate 300-fold higher than that of conventional WO₃/Cu₂O. This work provides a universal paradigm for designing Z-scheme systems with mismatched components, unlocking new possibilities for solar energy conversion and organic synthesis.

Introduction

Photocatalytic systems, which harness solar energy to drive redox reactions, hold immense potential for sustainable energy conversion and organic synthesis. 1-5 However, their efficiency is fundamentally limited by rapid charge recombination and inadequate redox potentials.^{6,7} Heterojunction photocatalysts, particularly step-scheme (S-scheme) systems, have emerged as a promising strategy to enhance charge separation while preserving strong redox capabilities (Scheme 1). 8-17 In S-scheme heterojunctions, the band bending, a built-in electric field and Coulombic attraction at the interface between oxidation (OP) and reduction photocatalysts (RP) promote selective recombination of low-energy carriers, retaining high-energy electrons and holes for surface reactions. 18-25 However, the construction of S-scheme heterojunctions for photocatalytic applications encounters two primary challenges. Firstly, optimizing the bandgaps is critical. To facilitate the step-scheme of photogenerated carriers in heterojunctions, both components must be simultaneously photo-excited.^{26,27} However, selecting

semiconductors to ensure a visible light response may compromise the overall redox capability of the S-scheme heterojunction photocatalytic system. Secondly, work function (Φ) matching between components is essential.²⁸⁻³⁰ The formation of an S-scheme heterojunction mandates that the work function of the RP is lower than that of the OP $(\Phi_{RP} < \Phi_{OP})$. These challenges severely limit material selection and often exclude semiconductors with superior light absorption or catalytic activity.

WO₃ and Cu₂O, as visible-light-responsive OP and RP candidates, exemplify this dilemma and have been excluded as the candidates for S-scheme heterojunction. $^{31\text{-}33}$ WO $_{\!3}$ possesses a highly positive valence band ideal for oxidation;³⁴⁻⁴⁰ while Cu₂O offers a negative conduction band suitable for reduction.41-44 However, their intrinsic work function mismatch $(\Phi_{WO3} < \Phi_{Cu2O})$ leads to the formation of undesired II-type heterojunction other than S-scheme heterojunction, where electrons accumulate in CB of WO₃ and holes in VB of Cu₂O, drastically diminishing their redox potentials. Previous attempts to address this challenge, such as doping or morphology tuning, have failed to reconcile interfacial energetics with efficient charge dynamics, leaving the core issue of Φ-mismatch unresolved.

Herein, we propose an interfacial engineering strategy to override the work function limitation by introducing a nitrogendoped carbon (N-C) mediator and metallic Cu nanoparticles at the WO₃/Cu₂O interface. The N-C layer, with a low Φ, creates a directional electron transfer pathway from N-C to WO₃, while Cu nanoparticles act as hole reservoirs for Cu2O, collectively

^{a.} Jiangsu Key Laboratory of Green Synthetic Chemistry for Functional Materials, Department of Chemistry, School of Chemistry & Materials Science, Jiangsu Normal University, Xuzhou, 221116, P. R. China.

b. State Key Laboratory of Precision Spectroscopy, East China Normal University, Shanahai, 200241, P. R. China,

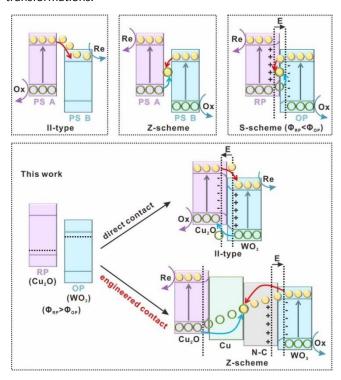
^{c.} Shenzhen HUASUAN Technology Co., Ltd, Shenzhen, 518055, P. R. China.

[†] These authors contributed equally.

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reshaping the built-in electric field to form a Z-scheme heterojunction. Through a "post-deposition and pyrolysis" synthetic approach, we constructed a tightly integrated WO₃/N-C/Cu/Cu₂O heterojunction, where transient absorption and insitu electron paramagnetic resonance data confirm that photogenerated electrons from WO₃ transfer to N-C and recombine with holes from Cu₂O on Cu nanoparticles via the N-C/Cu interface, preserving the high-energy carriers for redox reactions. The optimized system achieves a 99% yield in terminal alkyne homo-coupling and a hydrogen evolution rate 300-fold higher than conventional WO₃/Cu₂O under the illumination of blue light LED. This work not only resolves the long-standing Φ-mismatch challenge but also establishes a universal interfacial engineering paradigm for constructing Zscheme systems from incompatible components. By decoupling material selection from intrinsic electronic constraints, our strategy opens new avenues for developing high-performance photocatalysts tailored for solar fuel production and organic transformations.



Scheme 1. The schematic diagram of the migration and separation pathways of photogenerated carriers in II-type, Z-scheme, and S-scheme heterojunctions, as well as the heterojunctions developed in this work.

Results and discussion

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The synthesis procedure of mesoporous $WO_3/N\text{-}C/Cu/Cu_2O$ cuboctahedra via the "post-deposition and pyrolysis" method is schematically illustrated in Fig. 1a. Firstly, well-defined HKUST-1 with a cuboctahedral structure was synthesized through a simple solvothermal reaction, utilizing copper nitrate trihydrate $(Cu(NO_3)_2\cdot3H_2O)$ as metal salt, 1,3,5-benzenetricarboxylic acid as organic ligand, and benzimidazole as the modulator. For detailed experimental procedures, please refer to the ESI. Then,

the obtained HKUST-1 precursor and monoclinic. WO2 the relevant preparation and characterization details are presented in ESI and Fig. S1) were uniformly immersed in methanol. Subsequently, the mixture was heated at 120 °C for 6 hours. During this treatment, the cuboctahedra WO3/HKUST-1 precursor could be obtained. Afterward, a pyrolysis treatment was carried out at 300 °C in an Ar atmosphere, and Cu²+ ions within the WO3/HKUST-1 precursor were transformed into Cu2O and metallic Cu. Simultaneously, benzimidazole and 1,3,5-benzenetricarboxylic acid underwent in-situ carbonization to form an external nitrogen-doped carbon (N-C) layer. As a result, the WO3/HKUST-1 precursor evolved into mesoporous cuboctahedral WO3/N-C/Cu/Cu2O.

The composite and structure of the products obtained at different stages were characterized by powder X-ray diffraction (XRD) and field-emission scanning electron microscopy (FE-SEM). The XRD pattern of HKUST-1 precursor is shown in Fig. S2. The precursor exhibits sharp diffraction peaks in good agreement with the simulated pattern of HKUST-1 with high crystallinity.45 The FE-SEM image (Fig. 1b) illustrates that the prepared HKUST-1 precursor possesses a cuboctahedral morphology characterized by high uniformity and smooth surface. After the deposition of WO₃, the resulting products (Fig. 1c) maintain a uniform cuboctahedral structure with excellent monodispersity, and their surfaces are decorated with small particles. As revealed by XRD analysis (Fig. S3a), the diffraction peaks are well consistent with monoclinic WO₃ phase (PDF no. 01-072-1465) and simulated HKUST-1 pattern, thereby confirming the coexistence of WO₃ and HKUST-1 in the composite sample WO₃/HKUST-1. The simultaneous presence of Cu, W, O, C and N elements in WO₃/HKUST-1 was revealed by the energy-dispersive X-ray spectroscopy (EDX) elemental mapping (Fig. S3b-h). After annealing treatment in an Ar atmosphere, the WO₃/HKUST-1 phase transformed into WO₃/N-C/Cu/Cu₂O. The XRD pattern shown in Fig. S4 demonstrates a good match between the diffraction peaks of the annealed product and monoclinic WO₃ (PDF no. 01-072-1465), cubic Cu₂O (PDF no. 01-077-0199), and cubic Cu (PDF no. 00-004-0836). Raman spectrum of WO₃/N-C/Cu/Cu₂O (Fig. S5) shows a D-band peak at 1344 cm⁻¹ and a G-band peak at 1562 cm⁻¹, which are related with the defected graphic carbon. These results provide strong evidence for the successful phase transformation to WO₃/N-C/Cu/Cu₂O. The WO₃/N-C/Cu/Cu₂O composite have maintained the cuboctahedral morphology of WO₃/HKUST-1 precursor, but underwent a structural change from a solid to a hollow configuration, as depicted in Fig. 1d. The porosity of the obtained WO₃/N-C/Cu/Cu₂O composite was characterized using nitrogen adsorption-desorption isotherms (Fig. S6). The Barrett-Joyner-Halenda (BJH) pore sizes distribution has been determined to be approximately 19 nm, and the Brunauer Emmett Teller (BET) surface area of WO₃/N-C/Cu/Cu₂O has been measured to be around 61.6 m²/g.

The structural and component information of WO₃/N-C/Cu/Cu₂O was comprehensively investigated using transmission electron microscopy (TEM). The TEM image (Fig. 1e) reveals that the synthesized WO₃/N-C/Cu/Cu₂O exhibits a hollow cuboctahedral morphology, accompanied by dispersed

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small particles. The magnified TEM image (Fig. 1f) clearly reveals the presence of a well-defined heterojunction interface between the dispersed particles and the amorphous carbon layers. The HRTEM (high-resolution TEM) image of the marked yellow zone of the dispersed particle in Fig. 1f is depicted in Fig. 1g. The lattice spacing of 0.365 nm can be ascribed to the (200) plane of the monoclinic WO₃ phase, thereby indicating that the dispersed particle is a WO₃ particle. The magnified TEM image (Fig. 1h, red zone in Fig. 1e) indicates that the shell structure of the hollow cuboctahedra is assembled by nanoparticles. The HRTEM image of zone 1 in Fig. 1h (Fig. 1i) exhibits a lattice spacing of 0.208 nm, corresponding to the (111) plane of cubic Cu phase. The HRTEM image of zone 2 in Fig. 1h (Fig. 1j) displays a lattice spacing of 0.212 nm, which can be attributed to the (200) plane of cubic Cu₂O phase. The presence of the Cu/Cu₂O heterojunction interface can be confirmed by these findings. The high-angle annular dark-field scanning transmission electron microscope (HAADF-STEM) images in Fig. 1k further verify the existence of a hollow structure with small particles on the surfaces. The energy-dispersive X-ray spectroscopy (EDX) mapping (Fig. 1l-p) reveals a homogeneous distribution of Cu, W, O, C, and N elements within the cuboctahedral framework. The combination of XRD, HRTEM, and EDX analyses provides robust evidence for the successful synthesis of hollow cuboctahedra featuring a WO₃/N-C/Cu/Cu₂O interface.

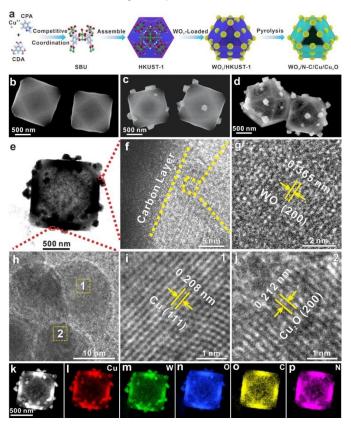


Fig. 1 (a) Schematic showing the synthetic process of WO₃/N-C/Cu/Cu₂O, FE-SEM images of (b) cuboctahedra HKUST-1 precursor, (c) cuboctahedra WO₃/HKUST-1 precursor, (d) WO₃/N-C/Cu/Cu₂O, (e) TEM image of WO₃/N-C/Cu/Cu₂O, (f) the magnified TEM image of marked area in Fig. 1e, (g) the corresponding HRTEM of marked area in Fig. 1f, (h) the magnified TEM image of marked area in Fig. 1e, (i, i) the corresponding HRTEM images

of area 1 and 2 in Fig. 1h, (k) STEM image of WO_3/N -C/Cu/Cu₂O, ($I_{\tiny T}$) EDX elemental mapping of Cu, W, O, C and N elements. DOI: 10.1039/DSSC05362E

The synthesized WO₃/N-C/Cu/Cu₂O heterojunctions were further analyzed using X-ray photoelectron spectroscopy (XPS) to determine the chemical states and static electron transfer between heterojunctions (Fig. S7 and Fig. 2a-d). To explore the electronic interactions among the components of the WO₃/N-C/Cu/Cu₂O heterojunction, WO₃/Cu₂O (details in ESI and Fig. S8), N-C/Cu/Cu₂O (details in ESI and Fig. S9), WO₃, and Cu₂O (details in ESI and Fig. S10) were employed as reference samples. As illustrated in Fig. 2a, the characteristic peaks located at 35.5 eV and 37.7 eV in the W 4f XPS spectrum of pristine WO₃ are attributed to the W 4f_{7/2} and W 4f_{5/2} of W⁶⁺, respectively.46 Compared with pristine WO₃, the W 4f peaks of the WO₃/N-C/Cu/Cu₂O heterojunction have shifted to a lower energy region, indicating that WO₃ in this heterojunction is in an electron-acquiring state. In contrast, WO₃ in WO₃/Cu₂O heterojunction is in the opposite state since the W 4f peaks have shifted to a higher energy region relative to pristine WO₃. In Fig. 2b, the peaks at binding energies of 932.8 eV and 952.6 eV are assigned to Cu 2p_{3/2} and Cu 2p_{1/2} of Cu⁺/Cu⁰ in the N-C/Cu/Cu₂O heterojunction, respectively. Meanwhile, the remaining peaks (934.8 eV and 954.7 eV) can be assigned to Cu²⁺, which is due to the surface oxidation of Cu⁺ ions and Cu⁰. In comparison with N-C/Cu/Cu₂O, the Cu 2p peaks in the WO₃/N-C/Cu/Cu₂O heterojunction shift towards a higher energy region. Combining with the previous analysis of the W 4f XPS spectra, this shift indicates that the directed transfer of interfacial electrons to WO₃ from N-C/Cu/Cu₂O during the formation of the WO₃/N-C/Cu/Cu₂O heterojunctions. While, Fig. 2c indicates that the characteristic peaks of Cu⁺ in the WO₃/Cu₂O heterojunction shift towards a lower binding energy, and no characteristic peaks of Cu2+ are detected. This observation indicates that in WO₃/Cu₂O heterojunction, interfacial electrons transfer from WO₃ to Cu₂O, supported by the above analysis of the W 4f XPS spectra. Fig. 2d presents the XPS spectra of C 1s, N 1s, and O 1s in the WO₃/N-C/Cu/Cu₂O heterojunction. In the C 1s XPS spectra, the peak at 284.8 eV corresponds to alkyl-C bonds, the peak at 286.1 eV is associated with sp²-carbon containing nitrogen atoms (C-N bonds), and the peak at 288.7 eV is assigned to carbon atoms bonded to carbonyl groups (C=O). The N 1s peak at 399.6 eV can be attributed to graphene-N, further confirming N doping within the carbon layer. The high-resolution O 1s spectrum can be deconvoluted into three major peaks centered at 530.7 eV, 532.0 eV, and 533.6 eV, corresponding to lattice oxygen (O_L), oxygen-deficient regions (O_V), and dissociated and chemisorbed oxygen (O_c), respectively. The analysis of XPS spectra not only clarify the chemical states and elemental compositions within the synthesized WO₃/N-C/Cu/Cu₂O heterojunction but also indicate an interesting phenomenon, i.e., the direction of interfacial electron migration is opposite in WO₃/N-C/Cu/Cu₂O and WO₃/Cu₂O heterojunctions.

To investigate the reason for the opposite direction of interfacial electron migration, we calculated the work functions (Φ) of WO₃, N-C layer, Cu₂O, and Cu, since work function is a

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critical parameter for understanding interfacial charge transfer mechanisms.⁴⁷ As shown in Fig. 2e-h, the work functions of WO₃, N-C layer, Cu₂O, and Cu have been calculated to be 4.54 eV, 3.55 eV, 4.71eV, and 4.69 eV, respectively. The work function of WO_3 is lower than Cu_2O , which facilitates the migration of interfacial electrons from WO₃ to Cu₂O when forming WO₃/Cu₂O heterojunction upon direct contact. Consequently, an internal electric field directed from Cu₂O to WO_3 is established at the WO_3/Cu_2O interface. In contrast, when WO₃ is in direct contact with the N-C layer, the lower work function of the N-C layer than WO₃ leads to the electron transfer from N-C layer to WO₃ at the interface, resulting in a built-in electric field oriented from the N-C layer toward WO3. The differences in work functions among WO₃, the N-C layer, and Cu₂O offer a reasonable explanation for the opposing directions of interfacial electron migration observed during the formation of the WO₃/N-C/Cu/Cu₂O and WO₃/Cu₂O heterojunctions. The distinct directions of electron migration at these interfaces results in oppositely oriented built-in electric fields, which underpin the II-type transfer mechanism of photogenerated carriers in the WO₃/Cu₂O heterojunction and the Z-scheme transfer mechanism of photogenerated carriers in the WO₃/N-C/Cu/Cu₂O heterojunction.

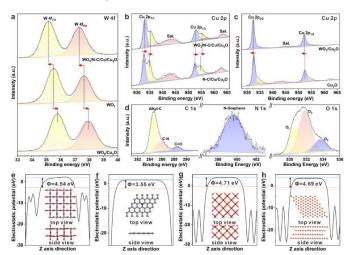


Fig. 2 (a) W 4f XPS of WO_3/N -C/Cu/Cu₂O, N-C/Cu/Cu₂O and WO_3 , (b) Cu 2p XPS of WO_3/N - $C/Cu/Cu_2O$ and $N-C/Cu/Cu_2O$, (c) Cu~2p~XPS of WO_3/Cu_2O and Cu_2O , (d) C~1s, N~1s, and O~1s1s XPS of WO₃/N-C/Cu/Cu₂O. The electrostatic potentials and corresponding models of (e) WO_3 , (f) N-C, (g) Cu_2O and (h) Cu. The red, light blue, grey, blue, and orange spheres represent O, W, C, N, and Cu atoms, respectively. Blue and red dashed lines indicate the

UV-vis diffuse reflectance spectra and valence band XPS spectra (VB-XPS) have been measured to investigate the optical properties and the band positions of the catalysts. As shown in Fig. S11a, the optical absorption edges of pristine WO₃ and Cu₂O fall within the visible light spectrum, thereby endowing the WO₃/N-C/Cu/Cu₂O heterojunction with excellent light response property in this region. The intrinsic band gap values (Eg) of pristine WO₃ and Cu₂O are estimated to be 2.54 eV and 1.82 eV, respectively (Fig. S11b). The valence band potential (E_{VB}) can be evaluated by VB-XPS spectra. According to Fig. S11c, the E_{VB} values of pristine WO₃ and Cu₂O are measured to be 2.62 V and 0.41 V (vs. NHE), respectively. The conduction band potential

(E_{CB}) can be calculated using the equation of VEGETERRED TO COMPARISON OF COMPARISON Therefore, the E_{CB} values for WO₃ and Cu₂O are estimated to be 0.08 V and -1.41 V (vs. NHE), respectively.

Electron paramagnetic resonance (EPR) measurements have also been conducted to systematically study the mechanisms of charge separation and transfer in WO₃/N-C/Cu/Cu₂O heterojunction. Superoxide $(\cdot O_2^-)$ and hydroxyl $(\cdot OH)$ radicals can be trapped by 5,5-dimethyl-1-pyrroline N-oxide (DMPO), a well-established spin-trapping agent, and subsequently detected through EPR spectroscopy. As illustrated in Fig. 3a and c, no signal can be detected under dark condition. Under 15 min of blue LED irradiation, the characteristic peaks of DMPO-O2and DMPO-·OH have been detected in the WO₃/N-C/Cu/Cu₂O heterojunction, while no such signals can be detected in the WO₃/Cu₂O heterojunction. Based on the aforementioned information, it is evident that the CB position of Cu₂O is more negative than the $O_2/\cdot O_2$ potential (-0.33 V vs. NHE), while the VB position of WO₃ is more positive than the H₂O/·OH potential (2.38 V vs. NHE). In contrast, the CB position of WO₃ is more positive than the $O_2/\cdot O_2$ potential, and the VB potential of Cu_2O is more negative than the H₂O/·OH potential. Therefore, the simultaneous generation of $\cdot O_2^-$ and $\cdot OH$ under light irradiation indicates that when WO₃/N-C/Cu/Cu₂O heterojunction as the catalyst to participate the reaction, the photogenerated electrons are ultimately enriched in the CB of Cu₂O, while the photogenerated holes are accumulated in the VB of WO₃. This provides strong evidence for the Z-scheme transfer mechanism in the WO₃/N-C/Cu/Cu₂O heterojunction. Neither ·O₂- nor ·OH is produced in WO₃/Cu₂O under light irradiation, consistent with the II-type charge transfer mechanism in WO₃/Cu₂O heterojunction. In addition, as exhibited in Fig. 3b and d, the peak intensities of $\cdot O_2^-$ and $\cdot OH$ radicals in $WO_3/N-C/Cu/Cu_2O$ gradually increases with prolonged in-situ irradiation time. This indicates that the Z-scheme structure facilitates efficient separation and transfer of photogenerated carriers, thereby promoting the generation of more active radical species. The above results from EPR and XPS confirm that inserting the N-C layer and Cu into the interfaces of WO₃/Cu₂O heterojunction enables the formation of a Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction.

The Z-scheme transfer mechanism of photogenerated carriers in WO₃/N-C/Cu/Cu₂O heterojunction under light illumination is further illustrated in Fig. 3e. When all the catalysts (WO₃, N-C layer, Cu, and Cu₂O) are in contact, three distinct interfaces are formed. Given that the N-C layer has a lower work function than WO₃, electrons spontaneously transfer from the N-C layer to WO₃ until the Fermi level at the interface reach equilibrium. This electron redistribution at the WO₃/N-C interface results in band bending and a build-in internal electric field near the interface. Due to the similar work functions of Cu and Cu₂O, they can form an ohmic contact directly. Upon illumination, WO₃ and Cu₂O become excited and generate electron-hole pairs. Under the influence of the internal electric field and band bending, the photogenerated electrons from the CB of WO₃ are driven towards the N-C layer. Due to the ohmic contact and the p-type characteristic of Cu₂O, the photogenerated holes from the VB of Cu₂O transfer to Cu

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while simultaneously consuming photogenerated electrons on the N-C layer. Consequently, the oxidation-type photocatalyst of WO₃ forms hole-rich regions, whereas the reduction-type photocatalyst of Cu₂O forms electron-rich regions. This process effectively enhances the transport of photogenerated charge carriers and maintains the high redox capability in the WO₃/N-C/Cu/Cu₂O heterojunction.

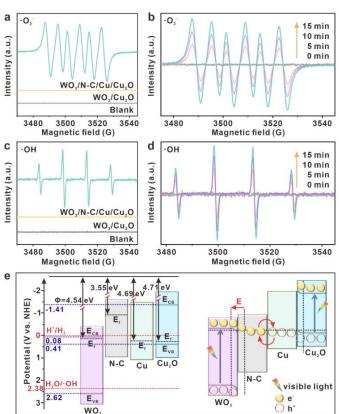


Fig. 3 EPR spectra of DMPO-·O₂- (a) in the presence of WO₃/N-C/Cu/Cu₂O and WO₃/Cu₂O heterojunctions upon irradiation with blue LEDs for 15 min and (b) in the presence of WO_3/N -C/Cu/Cu₂O upon irradiation with blue LEDs for 0, 5, 10 and 15 min. EPR spectra of DMPO- \cdot OH (c) in the presence of WO $_3$ /N-C/Cu/Cu $_2$ O and WO₃/Cu₂O heterojunctions upon irradiation with blue LEDs for 15 min and (d) in the presence of $WO_3/N-C/Cu/Cu_2O$ upon irradiation with blue LEDs for 0, 5, 10 and 15 min. (e) The Z-scheme charge transfer mechanism in WO₃/N-C/Cu/Cu₂O under visible light irradiation.

The charge transfer dynamics of the Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction has been investigated femtosecond transient absorption (fs-TA) spectroscopy. As shown in Fig. 4a, pristine WO₃ exhibits a pronounced excitedstate absorption (ESA) signal ($\Delta A>0$) at 750 nm. According to previous research reports,58 this signal is attributed to defecttrapped electrons. Notably, after constructing the WO₃/N-C/Cu/Cu₂O heterojunction, the ESA signal at 750 nm completely has vanished and is replaced by ground-state bleaching (GSB, ΔA<0) signal. This signal inversion demonstrates that interfacial charge transfer efficiently quenches defect-trapped electrons in WO₃. Combined with the energy structure analysis (Fig. 3e), it is evident that the conduction band (CB) electrons of WO₃ preferentially recombine with the valence band (VB) holes of Cu₂O through the N-C/Cu interfacial pathway, mediated by the N-C layer. This process blocks the accumulation of defect-state

electrons in WO3, thereby preserving the excited-state electrons in the CB of Cu2O for prolonged activations (>ንውን ያ These results corroborate our prior detection of long-lived GSB signals in Cu₂O-carbon composites when illuminated under near-infrared irradiation.50

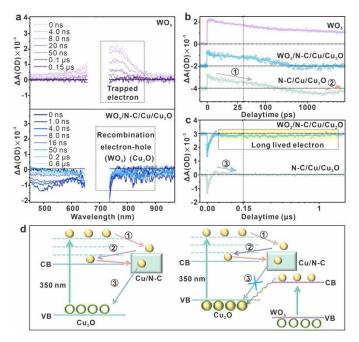


Fig. 4 (a) Transient absorption spectra of WO₃/N-C/Cu/Cu₂O and WO₃. (b, c) Kinetics of WO₃/N-C/Cu/Cu₂O, N-C/Cu/Cu₂O and Cu₂O at 750 nm. (d) Schematic diagram of charge migration paths in N-C/Cu/Cu₂O and WO₃/N-C/Cu/Cu₂O.

Fig. 4b and c show the kinetic curves at 750 nm and the data are fitted using multi-exponential decay components, as listed in Table S1 and S2. For pristine WO₃, a fast ESA decay (τ_1 =36 ps) has been provided, consistent with shallow defect electron trapping at surface defect states. Then the excited electron was trapped by deep-defect states with a time constant of about 37 ns. In contrast, the kinetic curve of N-C/Cu/Cu₂O was fitted by a multiexponential function with a non-decay offset. The optimal fit reveals two decay components: a fast one with a time constant of 6.0 ps and a slow one of 97 ps. The fast decay process is assigned to the trapping of holes in surface defects, while the 97 ps time constant should reflect the movement of electrons to deep-trapped states (process 1 in Fig. 4d). The 1.1 ns time constant reflects carrier de-trapping, which drives the formation of GBS (process 2 in Fig. 4d). The GSB recovers within 18 ns and is followed by an ESA signal with a time constant of 48 ns (process 3 in Fig. 4d). Notably, the heterojunction (WO₃/N-C/Cu/Cu₂O) displays accelerated decay processes with the time constant of 2.6 ps and slow 153 ps. The absence of detrapping process in WO₃/N-C/Cu/Cu₂O suggests interfacial charge transfer from WO₃ to Cu₂O. Then, the kinetic behavior of the heterojunction demonstrates a dual-phase evolution: GSB recovery (Table S2, τ_1 =3.0 ns, τ_2 =19 ns) followed by an ESA (τ₃=36 ns) signal, evidencing defect-engineered carrier lifetime extension via N-doped carbon layer. Remarkably, the WO₃/N-C/Cu/Cu₂O heterojunction shows unique behavior: following the initial GSB recovery, a secondary bleach feature grows

stronger, with carrier lifetimes surpassing 1 μ s. This divergence originates from the rapid interfacial recombination of conduction band (CB) electrons in WO₃ with valence band (VB) holes in Cu₂O at the heterojunction interface (Fig. 3e), thereby suppressing the charge transfer pathway (process 3) observed in N-C/Cu/Cu₂O. Consequently, the CB electrons in Cu₂O remain energetically active, significantly enhancing catalytic performance through interfacial carrier engineering.

To explore the role of the WO₃/N-C/Cu/Cu₂O interface in photocatalytic performance, a series of heterojunctions with various W-to-Cu molar ratios have been synthesized. The samples labelled as C/Cu, W/C/Cu-1, W/C/Cu-2. W/C/Cu-3, and W/C/Cu-4 represent the WO₃/N-C/Cu/Cu₂O heterojunctions with W-to-Cu molar ratio of 0, 1:3, 1:2, 2:3, and 3:4, respectively. As shown in the SEM images in Fig. 5a, as the Wto-Cu molar ratio increases, the density of WO₃ particles deposited on the surface of the N-C/Cu/Cu₂O cuboctahedra The homo-coupling significantly. reaction phenylacetylene to 1,4-Diphenylbutadiyne has been chosen as the target reaction (Fig. S12). As illustrated in Fig. 5a and b, the deposition of WO₃ to form a WO₃/N-C/Cu/Cu₂O interface enhances the photocatalytic activity of N-C/Cu/Cu₂O. The photocatalytic performance of the $WO_3/N-C/Cu/Cu_2O$ heterojunction improves with the addition of WO₃ until the Wto-Cu molar ratio reaching 3:4. Specifically, the photocatalytic activity of W/C/Cu-4 is lower than W/C/Cu-3. This decrease is mainly attributed to the excessive loading of WO₃, which obstructs the photoexcitation of Cu₂O and disrupts the smooth operation of the Z-scheme photogenerated carrier migration mechanism. Fig. 5c illustrates the comparative photocatalytic activities of WO₃/N-C/Cu/Cu₂O (with a W-to-Cu molar ratio of 2:3), N-C/Cu/Cu₂O, WO₃/Cu₂O, Cu₂O, and WO₃ in the terminal alkyne homo-coupling reaction. The results indicate that the photocatalytic activity follows the order: WO₃/N-C/Cu/Cu₂O > $N-C/Cu/Cu_2O > WO_3/Cu_2O > Cu_2O > WO_3$. Based on the reaction mechanism of terminal alkyne homo-coupling, 51-56 pristine WO₃ cannot generate the crucial Cu(I)-related intermediates and therefore exhibits no catalytic activity for this reaction. The catalytic performance of N-C/Cu/Cu₂O and WO₃/Cu₂O heterojunctions surpasses pristine Cu₂O due to the directional charge transfer at the heterojunction interface, which facilitates the conversion of Cu(I)-related species into electron-deficient Cu(II)-related species. This conversion enables the dissociation of Cu(II)-related species, leading to the formation of $C_{sp}\text{-}C_{sp}$ homo-coupling products.⁴⁸ Furthermore, the photocatalytic efficiency of the WO₃/N-C/Cu/Cu₂O heterojunction is significantly enhanced compared to N-C/Cu/Cu₂O and WO₃/Cu₂O. It benefits from both the internal charge rearrangement within the Cu(I)-related species and the generation of $\cdot O_2^-$, which further promotes the transformation into Cu(II)-related species. Since there is no obvious relationship between the surface area and the activity (Fig. S13), the significantly enhanced photocatalytic activity of WO₃/N-C/Cu/Cu₂O for the terminal alkyne homo-coupling reaction, as compared to WO₃/Cu₂O, offers compelling evidence supporting the Z-scheme migration mechanism within the WO₃/N-C/Cu/Cu₂O heterojunction. In addition, we have conducted

experiments to prove the important influences of the hotogenerated hole and electron, and $\cdot O_2$ (Fig. 105147). The order stability of the catalysts has been evaluated (Fig. 5d), and the WO₃/N-C/Cu/Cu₂O heterojunction demonstrates outstanding cycle stability with no obvious loss of photocatalytic efficiency and no changes in the structure and morphology after three cycles of photocatalytic reactions (Fig. S15). To further validate the high catalytic activity of WO₃/N-C/Cu/Cu₂O, we have conducted the photocatalytic oxidation of various alkynes substrates under identical conditions. As shown in Fig. 5e, the yields of the corresponding target products ranged from 80% to 99%, confirming that the Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction is an exceptional photocatalyst under blue light (λ =450 nm).

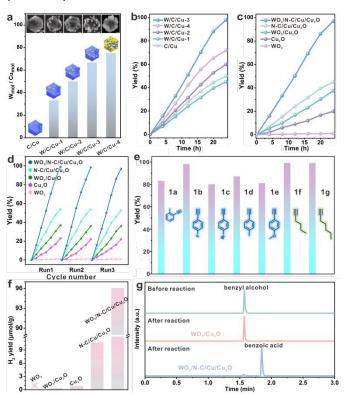


Fig. 5 (a) The histogram of different W/Cu contents and the corresponding SEM images, (b) the yield of homo-coupling of phenylacetylen by different W/Cu contents samples, (c) the yield of homo-coupling of phenylacetylen by WO₃/N-C/Cu/Cu₂O, N-C/Cu/Cu₂O, Cu₂O, WO₃/Cu₂O, and WO₃ samples, (d) stability test of the various photocatalysts for the homo-coupling of phenylacetylene, (e) performance of WO₃/N-C/Cu/Cu₂O in homo-coupling of various alkynes, (f) the photocatalytic hydrogen yields over WO₃/N-C/Cu/Cu₂O, N-C/Cu/Cu₂O, Cu₂O, WO₃/Cu₂O, and WO₃ samples with benzyl alcohol as the hole capturer, (g) the retention time of different aqueous solution (unreacted benzyl alcohol solution and reaction solutions with WO₃/Cu₂O and WO₃/N-C/Cu/Cu₂O as photocatalysts, respectively) detected by liquid chromatography.

To further confirm the Z-scheme mechanism of the WO_3/N -C/Cu/Cu₂O heterojunction and its influence on catalytic performance, the WO_3/N -C/Cu/Cu₂O heterojunction has been employed as catalyst for photocatalytic hydrogen production using benzyl alcohol as the hole capture. As illustrated in Fig. 5f, under irradiation of blue light, the hydrogen production efficiency of the five samples follows the order: WO_3/N -C/Cu/Cu₂O > N-C/Cu/Cu₂O > VO_3/Cu_2 O >

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Amongst, the CB potential of WO₃ is more positive than the H⁺/H₂ potential (0 V vs. NHE), therefore WO₃ lacks the ability to photocatalyze hydrogen production. The hydrogen production rate of WO₃/N-C/Cu/Cu₂O is 9 times and 120 times higher than N-C/Cu/Cu₂O and Cu₂O, respectively, indicating that the constructed WO₃/N-C/Cu/Cu₂O heterojunction interface can significantly facilitate efficient migration and separation of photogenerated carriers. Moreover, the hydrogen production of WO_3/N -C/Cu/Cu₂O is 300 times greater than WO_3/Cu_2O , suggesting that the photogenerated electrons in WO₃/N-C/Cu/Cu₂O are predominantly accumulated in the CB of Cu₂O, thereby promoting the reduction reaction for hydrogen production. The reaction solutions after the photocatalysis over these five catalysts have been analyzed using liquid chromatography, and the results are presented in Fig. 5g and Fig. S16. With WO_3/Cu_2O , $N-C/Cu/Cu_2O$, and Cu_2O as photocatalysts, only unreacted benzyl alcohol can be detected, indicating that no conversion of benzyl alcohol to benzoic acid has occurred under these conditions. In contrast, in the reaction solution of WO₃/N-C/Cu/Cu₂O and WO₃, both the benzyl alcohol and benzoic acid have been observed. The benzoic acid can be produced via the oxidation of benzyl alcohol by OH or photogenerated holes. This observation can be attributed to the VB position of WO₃ (+2.62 V vs. NHE) being more positive than the redox potential of benzaldehyde/benzoic acid couple (+2.5 V vs. NHE).⁴⁹ These findings suggest that in the WO₃/Nphotogenerated holes ultimately C/Cu/Cu₂O system, accumulate on the VB of WO₃ to participate in the oxidation reaction, providing another strong evidence for the preserved high redox potentials in the obtained Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction.

To further verify the effective separation and migration of photogenerated carriers within the constructed Z-scheme WO₃/N-C/Cu/Cu₂O heterojunction, we have conducted measurements of photocurrent density and electrochemical impedance spectroscopy (EIS) for WO₃/N-C/Cu/Cu₂O, N-C/Cu/Cu₂O, WO₃/Cu₂O, Cu₂O, and WO₃ samples. As illustrated in Fig. S17a, the photocurrent densities of $WO_3/N\text{-}C/Cu/Cu_2O$ are 2, 3, 5, and 5 times higher than N-C/Cu/Cu₂O, WO₃/Cu₂O, Cu₂O, and WO₃, respectively. This suggests that the formation of multiple interfaces does not undermine the positive promoting effect of the WO₃/N-C/Cu/Cu₂O heterojunction on the separation of photogenerated carriers. The order of EIS radius of these five samples (Fig. S17b) is WO₃/N-C/Cu/Cu₂O < $N-C/Cu/Cu_2O < WO_3/Cu_2O < Cu_2O < WO_3$. Generally, a smaller arc radius implies higher charge transfer efficiency. Therefore, WO₃/N-C/Cu/Cu₂O exhibits the highest charge transfer efficiency among these five samples. This further confirms that WO₃/N-C/Cu/Cu₂O interface in the Z-scheme heterojunction we constructed is sufficiently robust, providing an efficient and smooth pathway for the migration and separation of photogenerated carriers.

Conclusions

In this work, a "post-deposition and pyrolysis" synthesis strategy has been employed to fabricate a Z-scheme WO₃/N-

C/Cu/Cu₂O heterojunction featuring a tightly integrated interface. The XPS spectra analysis Pand ୀ Wଡ଼ିମାନ୍ଦି ହେନ୍ତିନ calculation reveal the opposite directions of interfacial electron migration WO₃/N-C/Cu/Cu₂O and WO₃/Cu₂O heterojunctions. These findings preliminarily demonstrate that the modified heterojunction interface can effectively overcome the mismatched work function of WO₃ and Cu₂O, control the direction of ground-state electron migration, thereby modulating the orientation of the built-in electric field at the interface. The results of EPR and photocatalytic tests have confirmed that the WO₃/N-C/Cu/Cu₂O heterojunction can preserve the energetic photogenerated holes of WO₃ and the photogenerated electrons of Cu₂O for participation in the redox reaction. The fs-TA spectroscopy elucidates the migration pathway of photogenerated carrier in the WO₃/N-C/Cu/Cu₂O heterojunction. Specifically, the photogenerated electrons from WO₃ recombine with the photogenerated holes of Cu₂O enriched on Cu through the mediation of the N-C layer, which is a typical Z-scheme charge transfer mechanism. This research not only introduces a novel concept in interface engineering for constructing efficient heterojunction photocatalysts but also provides a synthetic strategy for building effective and compact heterojunction interfaces.

Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript. In detailed, H. Gao, X. He, and J. Li contributed equally in the investigation and data curation, C. Qin supported the visualization, S. Zhi supported the data curation, L. Yang and J. Zhao contributed to the software, L. Sun, W. Zhan, and X. Han contributed to the conceptualization and writing.

Conflicts of interest

There are no conflicts to declare.

Data availability

The data supporting this article have been included as part of the Supplementary Information.

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Data availability statements

The data supporting this article have been included as part of the Supplementary Information.