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## Oriented design and engineering of advanced metal–organic frameworks for light hydrocarbon separations

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Light olefins, such as ethylene ( $C_2H_4$ ) and propylene ( $C_3H_6$ ), are essential feedstocks for the production of chemical products. However, the current purification strategy of distillation is energy-intensive and results in high carbon emissions. Adsorptive separation, the selective capture of gas from mixtures by porous materials, is considered a promising alternative or transitional technology. Metal–organic frameworks (MOFs), a kind of porous material with highly tunable nature, have emerged as an innovative chemistry in the past two decades, offering solutions for separating these small gases. This review highlights recent advances in the design and engineering of advanced MOFs, with a focus on precise control over their pore structure and functionality for the adsorption-based purification of  $C_2H_4$  and  $C_3H_6$  from the corresponding hydrocarbons with the same carbon number. The importance of rational design in achieving specific functionalities, such as functional sites and molecular sieving in rigid MOFs and local/global dynamics in soft MOFs, is underscored, with examples demonstrating enhanced performance in selective adsorption separation. Additionally, methods and examples of large-scale synthesis of MOFs are briefly described. The goal is to present the state-of-the-art chemistry and applications of MOFs and to offer an outlook towards discovering and designing further new materials.

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### 1. Introduction

The global production and demand for light olefins, especially polymer-grade ethylene ( $C_2H_4$ ) and propylene ( $C_3H_6$ ), exceed

that of all other commodity organic compounds. The majority of these products are widely used in chemical manufacturing, such as plastics, synthetic rubber, and synthetic fibers.<sup>1</sup> Additionally, they are essential raw materials for agriculture and fine chemicals. Currently,  $C_2H_4$  and  $C_3H_6$  are isolated from the cracking units through the strategies of energy-intensive catalytic hydrogenation and cryogenic distillation methods, which account for nearly 3% of all separation energy.<sup>2,3</sup> Furthermore, the high pressure and low temperature bring significant risks during the separation process. Therefore, there is a strong need to develop energy-efficient separation technologies, such as

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Table 1 Properties of C<sub>2</sub> and C<sub>3</sub> hydrocarbons

Molecular dimensions (Å <sup>3</sup> )	Kinetic diameter (Å)	Polarizability (×10 <sup>-24</sup> cm <sup>-3</sup> )	Boiling point (K)
C <sub>2</sub> H <sub>2</sub>	3.32 × 3.34 × 5.70	3.3	3.33–3.93
C <sub>2</sub> H <sub>4</sub>	3.28 × 4.18 × 4.82	4.16	4.25
C <sub>2</sub> H <sub>6</sub>	3.81 × 4.08 × 5.70	4.44	4.43–4.47
C <sub>3</sub> H <sub>4</sub>	4.16 × 4.01 × 6.51	4.2	5.55
C <sub>3</sub> H <sub>6</sub>	4.65 × 4.16 × 6.44	4.6	6.26
C <sub>3</sub> H <sub>8</sub>	4.20 × 4.60 × 6.80	4.3//5.12	6.29–6.37

non-thermally driven processes, for downstream processing of cracked gas.<sup>4–6</sup>

Adsorption separation is often considered as an alternative technology due to its ability to separate gas molecules based on their chemical nature or size, rather than their boiling points. This non-thermal process can be carried out under mild conditions, such as at room temperature, resulting in significantly reduced energy inputs. However, the main challenge of this technology lies in the construction or selection of optimal porous materials that can achieve the desired uptakes and separation factors. This is particularly difficult due to the extremely similar molecular properties of light hydrocarbons, especially when dealing with mixtures containing the same number of carbon atoms (Table 1). This similarity in properties, both physical and chemical, makes the efficient separation of these molecules a highly challenging task.

Porous materials, such as zeolites and activated carbons, have long been studied for their potential in separating light hydrocarbons.<sup>7–10</sup> For example, silver ZK-5 zeolite, known for its unique pore structure and surface chemistry, has shown promise in separating C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>. However, the strong interaction between the host and guest molecules makes it difficult to desorb them at higher temperatures (>200 °C), limiting their practical applications.<sup>11</sup> Activated carbons, on the other hand, have an irregular pore system with meso- or macropores, leading to co-adsorption and making it challenging to obtain pure C<sub>2</sub>H<sub>4</sub>. In summary, these traditional porous

materials, lacking structural diversity and design flexibility, are not suitable for meeting the demands of industrial hydrocarbon separations.

Metal–organic frameworks (MOFs), also known as porous coordination polymers (PCPs), are a new class of crystalline porous materials, which are built from the coordination assembly of organic linkers and inorganic nodes in two or three dimensions.<sup>12–21</sup> MOFs are emerging porous materials due to their charming diversity, exceptional porosity, functional pore surface and tunable nature (pore size, shape and distribution).<sup>22–29</sup> Over the past two decades, there have been numerous reports on MOFs, with many designs specifically targeting the separation of light hydrocarbon mixtures (Scheme 1). These designs have led to significant improvements in key properties, such as capacity and selectivity (Table 2). Typically, as a kind of rigid framework with open Cu sites, HKUST-1 was investigated as the first MOF for the separation of C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> mixtures as early as 2002.<sup>30</sup> After modifying the isolated open metal sites to a linear configuration, Fe<sub>2</sub>(dobdc) (dobdc: 2,5-dioxido-1,4-benzenedicarboxylate) demonstrated a sharply promoted separation performance of C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub> at 318 K, as the linear Fe atoms exhibited a strong affinity towards C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub>.<sup>31</sup> Since the prediction of structural dynamics by Prof. Kitagawa in 1998,<sup>75</sup> active research shifted to design and synthesis of soft MOFs for such separation, given the structural sensitivity of these structures to the tiny difference in the light hydrocarbons.<sup>76</sup> With a rational pyridyl ring (also called



Chunze Yu

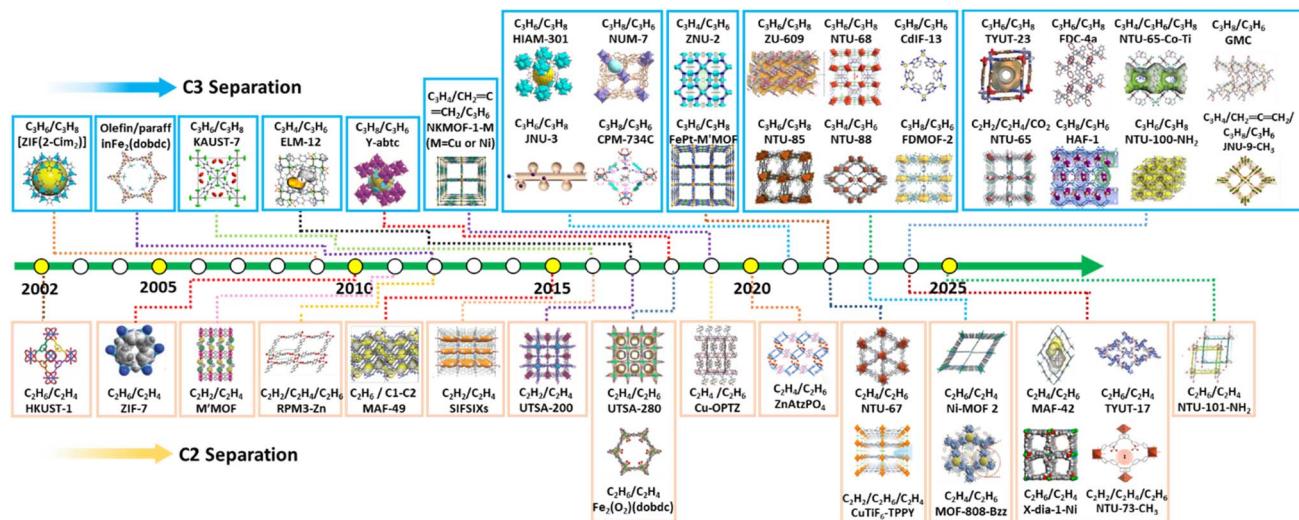
Chunze Yu graduated from Yili Normal University with a B.S. degree in 2024. In the same year, he joined Xinjiang University as a master's student under the supervision of Prof. Jingui Duan. His main interests are in the field of porous coordination polymers for efficient light hydrocarbon separation.



Jingui Duan

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Scheme 1 The representative MOFs for the separation of light hydrocarbons.<sup>6,30–74</sup>

a confined-rotational shutter) in the confined nanospace, NTU-88 creates a maximum aperture of 4.4 Å, allowing dedicated propyne C<sub>3</sub>H<sub>4</sub> (4.4 Å) adsorption from a C<sub>3</sub>H<sub>6</sub> (5.4 Å) containing mixture under ambient conditions.<sup>69</sup> Moreover, the dynamic nature of the molecular pockets in JNU-3a is pivotal for achieving molecular sieving of C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub>.<sup>65</sup> In this regard, MOFs have shown significant advantages over traditional materials in the separation of light hydrocarbons. Consequently, the present review outlines recent advances in MOF chemistry in the separation of light hydrocarbons, with a particular focus on the oriented design strategy, including functional sites in rigid MOFs, molecular sieving in rigid MOFs, and local dynamics and global flexibility in soft MOFs. The discussion will incorporate computational modelling and gas-loaded crystallographic structure to reveal the host–guest interaction and emerging functions in the most promising and innovative achievements. In addition, the strategies for large-scale MOF synthesis were also analysed. Separations of methane and carbon dioxide from light hydrocarbons are not included.

## 2. Design strategies

In the context of the continuous advancement of sophisticated MOFs for the separation of gases, a range of design strategies have been put forward. These can be broadly categorized into two groups based on their principles of gas molecular adsorption separation: (1) functional sites, molecular sieving and diffusion strategies using rigid MOFs, and (2) framework dynamics strategies in soft MOFs (Fig. 1). And, we compare the advantages and limitations of each strategy in order to provide clear suggestions and guidance for the synthesis of future MOFs (Table 3). Furthermore, the incorporation of host–guest interactions has been comprehensively examined in relation to these design strategies.

### 2.1 Oriented design strategies in rigid MOFs

**2.1.1 Open metal sites.** Open metal sites are defined as the vacant sites of metal ions or metal clusters within a metal

organic framework (MOF) that have unsaturated coordination numbers and are therefore open and reactive.<sup>77,78</sup> These sites have the capacity to coordinate or interact with gas molecules, organic molecules and ions. When metal sites are exposed in the pores or surfaces of the framework, they are susceptible to contact and react with molecules or ions in the external environment. The regulation of the types, quantities, and distributions of these sites can be achieved through the manipulation of the composition and structure of MOFs, thereby enabling selective recognition and separation of different molecules or ions.<sup>79</sup> This selective recognition capability renders MOFs promising candidates for application in the separation of light hydrocarbons.

Given the finding that HKUST-1 can separate C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub>,<sup>30</sup> the configuration of open metal sites has been investigated. This has been achieved by altering the transition metal ions in M-MOF-74 (M = Co, Mn, and Mg), resulting in varying abilities of the rod-shaped and high-density open metal sites to interact with C<sub>3</sub>H<sub>6</sub> through π-complexation.<sup>22</sup> It is worth noting that Co sites exhibit the largest difference in binding energies between C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub>. As pressure increases, the open Co sites become increasingly occupied by C<sub>3</sub>H<sub>6</sub>, leading to a strong suppression of C<sub>3</sub>H<sub>8</sub> adsorption and a significant increase in C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub> selectivity (Fig. 2a). Following the alteration of the redox-active Fe(II) within the same MOF platform, the material exhibits olefin adsorption selectivity that surpassed that of paraffin.<sup>31</sup> The neutron diffraction data confirmed that the unsaturated hydrocarbons such as C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub> exhibited side-on binding modes, with Fe–C distances of 2.42(2) to 2.60(2) Å. A comparison with the shorter distance of 2.020(5) to 2.078(4) Å observed in the diamagnetic complex [Fe(C<sub>2</sub>H<sub>4</sub>)<sup>4</sup>]<sup>2-</sup> indicated that the Fe centres within Fe-MOF-74 maintain a high spin electron configuration when binding these unsaturated gases. The interactions of both C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> with Fe in Fe-MOF-74 are even weaker, as evidenced by the longer Fe–C distance of ~3 Å (Fig. 2b).

In the context of the larger channel aperture of 11–12 Å in the MOF-74-series, a microporous framework, Fe(pyrazine)Pt(CN)<sub>4</sub>,



Table 2 Structural properties and adsorption separation performance of some key MOFs from Scheme 1

Typical MOFs	Structural properties	C2/C3 separation	Adsorption uptake (mmol g <sup>-1</sup> )	Selectivity	T/P	Ref.
HKUST-1 [ZIF(2-Cim <sub>2</sub> )]	3D channel with Cu sites	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	5.0/5.9 mol kg <sup>-1</sup>	—	295 K 0.8 bar	30
ZIF-7	SOD network	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub>	160/155 mg g <sup>-1</sup>	—	303 K 0.8 bar	45
M-MOF-3	SOD network	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	2.24/2.2	—	298 K 1 bar	38
MAF-49	Hexagonal network	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub>	6.56/1.35	—	195 K 1 bar	59
SIFSIXs-2-Cu-i	3D framework with narrow 1D zigzag channels	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	1.73/1.7	9	316 K 1 bar	49
KAUST-7	Cubic topology	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub>	4.02/2.19	44.54	298 K 1 bar	34
Y-3bte	3D framework with square-shaped channels	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub>	1.4/0.04	—	318 K 1 bar	32
UTSA-200	Flw network and cage-like pores	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub>	2.0/0.07	—	318 K 1 bar	6
UTSA-280	Doubly interpenetrated nets	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	3.65/0.63	6320	198 K 1 bar	44
Fe <sub>2</sub> (O <sub>2</sub> )(dobdc)	1D Ca-C <sub>4</sub> O <sub>4</sub> chain with pentagonal bipyramidal structure	C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub>	2.5/0.098	>10 000	298 K 1 bar	70
JNU-3	Iron-peroxy sites	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub>	3.32/2.53	4.4	298 K 1 bar	46
	1D diffusion channel with dynamic molecular pockets	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub>	58.6/42.5 cm <sup>3</sup> g <sup>-1</sup>	51.3	303 K 1 bar	65
ZnATzPO <sub>4</sub>	2D pillared framework with pocket-bottleneck channels	C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub>	1.1/0/3	31	273 K 1 bar	35
CuTiF <sub>6</sub> -TPPY	Pillared semi-cage 1D channels	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	3.62/2.82/2.42	—	298 K 1 bar	67
NTU-85	Square-apertured H <sub>2</sub> O channels	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub>	0.45/0.003	1.570	1.17 C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	—
NTU-88	SqI layer and rhombic pores	C <sub>2</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>6</sub>	86.0/2.0 cm <sup>3</sup> g <sup>-1</sup>	—	298 K 1 bar	37
FDMOF-2	3D framework	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub>	5.04/4.15	2.18	298 K 1 bar	69
CdHf-13	SOD network	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub>	2.56/2.32	2.04	298 K 1 bar	73
MOF-808Bzz	Octahedral morphologies	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	2.98/2.20/1.43	3.15 C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub>	298 K 1 bar	43
				1.90 C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	298 K 1 bar	53
				1.92 C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	298 K 1 bar	53
ZU-609	3D networks with large 1D channels	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub>	2.0	>10	298 K 1 bar	33
HAF-1	3D framework with channels and molecular pockets	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub>	101.61 cm <sup>3</sup> cm <sup>-3</sup> /—	1.67 × 10 <sup>7</sup>	298 K 1 bar	55
X-dia-1-Ni <sub>0.89</sub> Co <sub>0.11</sub>	Flexible diamondoid networks	C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub>	4.96/0.54	—	273 K 1 bar	56
TYUT-17	Spindle-like cages	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	67.4/61.3 cm <sup>3</sup> g <sup>-1</sup>	—	298 K 1 bar	66
NTU101-NH <sub>2</sub>	H-bond-tuned interpenetrated pcu framework	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub>	40.1/15.2 cm <sup>3</sup> g <sup>-1</sup>	—	328 K 0.5 bar	62
NRMOF-1-Ni	3D framework with 1D channels and dual gas-binding sites	C <sub>2</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> /C <sub>3</sub> H <sub>6</sub>	3.5/2.1/—	1271.6(C <sub>3</sub> H <sub>4</sub> )	298 K 1 bar	52
JNU-9-CH <sub>3</sub>	3D framework with cubane SBUs	C <sub>3</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> /C <sub>3</sub> H <sub>6</sub>	3.6/3.4/2.95/2.9	1.5 (C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub> ) 2.1 (C <sub>3</sub> H <sub>4</sub> /C <sub>3</sub> H <sub>6</sub> ) 1.3 (propadiene/C <sub>3</sub> H <sub>6</sub> )	298 K 1 bar	60

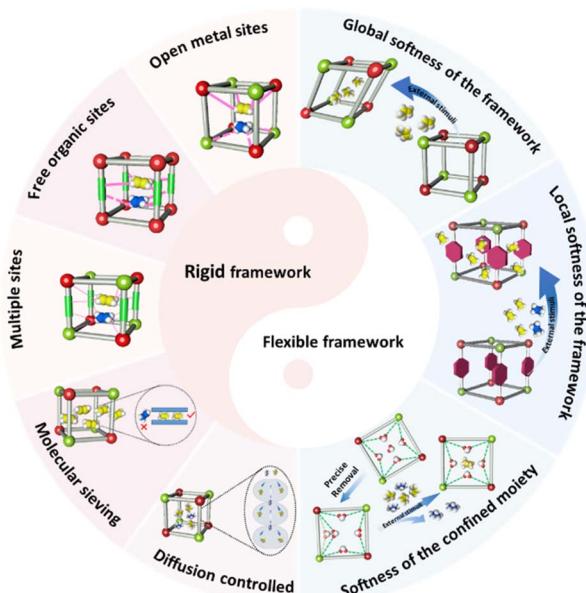


Fig. 1 Oriented design and engineering strategies of advanced MOFs.

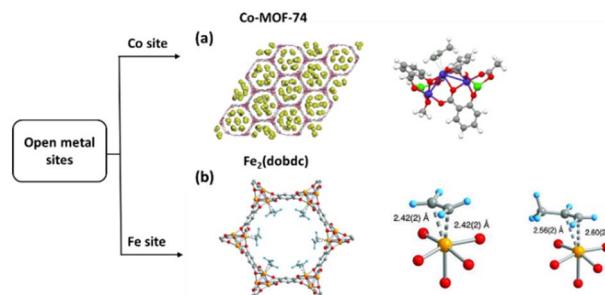


Fig. 2 (a) A snapshot from a GCMC simulation for  $\text{C}_3\text{H}_6$  adsorption in Co-MOF-74, and the propylene molecule binds to the Co atom. (b) A portion of the solid-state structure of  $\text{Fe}_2(\text{dobdc}) \cdot 2\text{C}_2\text{D}_4$ , and  $\text{C}_2\text{H}_4$  and  $\text{C}_3\text{H}_6$  coordination with  $\text{Fe}_2(\text{dobdc})$ . Reprinted with permission from ref. 22. Copyright 2012 John Wiley & Sons, ref. 31. Copyright 2012 The American Association for the Advancement of Science, respectively.

with a channel size of 4.0 Å and mixed open metal sites was designed and prepared for the separation of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$ .  $\text{C}_3\text{H}_6$  was located between two pyrazine rings, forming  $\pi-\pi$  interactions. Notably, the distance between  $\text{C}_{\text{C}_3\text{H}_6}$  and the face-to-face packed open Pt sites is short, ranging from 2.8–3.2 Å. In

Table 3 The advantages and limitations of each design strategy for light hydrocarbon separation-oriented MOFs

Design strategy	Advantages	Limitations
Open metal sites	High selectivity Highly adjustable High adsorption capacity	Synthesis and scale-up challenges Stability issues
Free organic sites	Highly selective Highly adjustable Good stability to avoid toxicity of metal sites Low regenerative energy consumption	Vulnerability to impurity gases Diffusion resistance
Multiple sites	Synergistically enhanced selectivity Broadening the applicable separation system Balancing adsorption strength and regeneration energy Improve stability	Difficulty and cost of synthesis Vulnerable to other impurity gases Selectivity may be low Synthesis complexity increase Risk of site-to-site interference Challenging kinetic balance
Molecular sieving	Highly selective separation Gentle operating conditions High stability (not dependent on chemisorption)	Aperture regulation is difficult Low pressure adsorption capacity Limited separation of similar-sized molecules Diffusion kinetic limitations
Diffusion controlled	Low energy consumption for regeneration Suitable for kinetic separations Highly dynamic selectivity Low energy consumption for regeneration High stability	Highly sensitive to an orifice structure. Low pressure separation efficiency Limited separation of similar sized molecules May be affected by the 'blocking effect' Structural stability issues High material preparation requirements
Global softness of the framework	Gentle operating conditions High selectivity and dynamic adaptability Low energy consumption and efficient regeneration Outfield response characteristics	Representation challenge Complex interaction mechanisms Difficult to synthesize Industrial scale-up challenges
Local softness of the framework	Highly selective Fast adsorption kinetics Good stability Precise molecular recognition Low energy consumption	Synthesis is complex Difficulty in industrial scale-up Regeneration difficulties
Softness of the confined moiety in the framework	Highly selective Energy saving potential	



contrast, the  $C_{C_3H_6}\cdots Ni$  distance becomes longer (3.5 Å) in FeNi-M' MOF. The finely tailored pore size of 4.0 Å facilitates the passage of  $C_3H_6$  through the channel, but significantly hinders the movement of the larger  $C_3H_8$  molecule.

**2.1.2 Free organic sites.** As an essential building block of MOFs, the functionalization of organic ligands is also expected to achieve efficient separation. Typically, the polar groups containing electronegative F and N elements are incorporated into the organic linker. In addition, amino and nitro groups, which act as electron donors or acceptors, have been investigated for their ability to selectively capture olefins from mixtures containing paraffins *via* supramolecular interactions. Interestingly, the metal-oxo species, formed by coordinating  $O_2$  to open metal sites have shown preferable interactions with paraffin. Furthermore, comparison of open metal site strategies may show reduced separation efficiency in the presence of moisture, and the use of organic sites not only achieves the desired separations, but also maintains good cycling stability even in the presence of moisture.

Supramolecular interactions derived from organic sites and light hydrocarbons are visualized and understood in the case of NOTT-300.<sup>80</sup> The results of neutron scattering, synchrotron X-ray and neutron diffraction, and computational modelling revealed that the free OH and benzene ring are simultaneously incorporated with  $C_2H_2$  through stronger hydrogen-bonding ( $C\cdots H = 2.96\text{--}3.64$  Å),  $\pi\cdots\pi$  stacking interactions ( $C\equiv C\cdots C_6 = 3.81$  Å) and intermolecular dipole interactions ( $C^I\cdots H^I = 3.24$  Å,  $C^II\cdots H^I = 3.12$  Å). However, the reduced  $\pi$ -electron density of  $C_2H_4$  compared to that of  $C_2H_2$  enables a longer  $C_{C_2H_4}\cdots HO$  distance (4.62 Å), which suggests a weaker hydrogen bonding. Despite the  $\pi\cdots\pi$  interactions between  $C_2H_4$  and phenyl rings of the ligand,  $C_2H_4$  was displaced towards the centre of the cavity and interacted majorly with  $C_2H_2$  molecules (captured by OH), upon the introduction of an equimolar mixture of the two gases to the system (Fig. 3a).

By changing the OH group into COOH, we recently found the unique ability for the adsorption of  $C_2H_2$  from  $C_2H_4$  containing mixtures by a MOF with crab-like carboxylic pincers. On a new platform of a 6-c topology network, pore size engineering of the carboxylic-functionalized MOFs was finely tuned by altering the  $MF_6^{2-}$  pillars ( $M = Si, Ti$  and  $Zr$ ).<sup>81</sup> Following the increased distance of the M-F, the pincer distance increased gradually from 5.02 to 5.91 Å, yielding a fitted nanospace in NTU-72 for selective recognition of a  $C_2H_2$  tetramer. The  $O_{COOH}$  interacts with terminal  $H_{C_2H_2}$  with a short distance of 2.309 Å, while the  $OH_{COOH}$  chelates to two  $C_2H_2$  through  $O\cdots H\cdots C_2H_2$ . The formation of a  $C_2H_2$  tetramer with  $C_{2h}$  symmetry is characterised by these two groups of gas molecules, which is distinctly different from the  $S_4$  symmetric  $C_2H_2$  tetramer in SIFSIX-Cu-i.<sup>34</sup> In contrast, two types of longer hydrogen bonds are observed between  $COOH\cdots C_2H_4$ , reflecting the relative weak carboxylic- $C_2H_4$  interaction. Importantly, the captured  $C_2H_2$  tetramer not only facilitates the direct harvesting of highly pure  $C_2H_4$  at the adsorption step, but also benefits the collection of pure  $C_2H_2$  during the desorption step (Fig. 3b).

To promote the ability for purification of  $C_2H_4$  from C2 ternary hydrocarbons, fine material design is highly required as

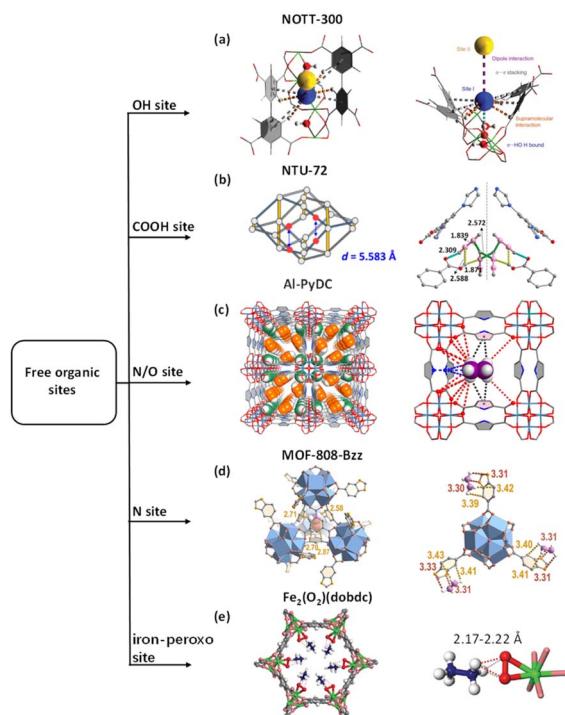


Fig. 3 (a) Schematic representation of the action of combining  $C_2H_2$  and  $C_2H_4$  molecules. (b) The pincer distance of NTU-72 and structure of  $C_2H_2$ -loaded NTU-72. (c) The SCXRD structure of  $C_2H_2$ -loaded Al-PyDC and the  $C_2H_6$  binding site in gas-loaded Al-PyDC. (d)  $C_2H_4$  binding sites I and sites II in MOF-808-Bzz. (e) Structure of  $Fe_2(O_2)(dobdc)_2C_2D_6$ . Reprinted with permission from ref. 80. Copyright 2015 Springer Nature Limited, ref. 81. Copyright 2023 The Royal Society of Chemistry, ref. 83. Copyright 2023 Springer Nature Limited, ref. 53. Copyright 2023 John Wiley & Sons, ref. 46. Copyright 2018 The American Association for the Advancement of Science, respectively.

the strength of the interactions between open metal sites and hydrocarbons typically increases in the order of  $C_2H_6 < C_2H_4 < C_2H_2$ . Based on the same 6-c platform, we replaced the functional group of COOH by  $-CH_3$  and  $CH_3$ .<sup>48</sup> The varied pore environment and organic sites endow NTU-73-CH<sub>3</sub> with remarkable capability for the direct production of poly-grade  $C_2H_4$  from ternary C2 hydrocarbons under ambient conditions. Comparably, the precursor framework of NTU-73-COOH cannot purify  $C_2H_4$ , while NTU-73-CF3 exhibits only negligible capability in this regard. This is primarily attributable to the modified binding sites in the NTU-73 series, which not only eliminate the channel obstruction caused by the formed  $C_2H_2$  tetramer, but also enhance the interactions of host- $C_2H_2/C_2H_6$ . These interaction changes were clearly observed by gas-loaded structures, and TJT-100 also demonstrates the ability for  $C_2H_4$  purification from ternary ( $C_2H_2/C_2H_6/C_2H_4$ ) mixtures by utilizing the abundant O sites.<sup>82</sup>

To further promote the host- $C_2H_6$  and host- $C_2H_2$  interactions, multiple supramolecular binding sites have been designed in the MOFs with suitable pore sizes. For example, Al-PyDC was assembled with ligands containing N and a large number of polar O sites, providing abundant supramolecular binding sites for  $C_2H_6$ , while the electronegative O and N sites



form hydrogen bonds with  $C_2H_2$ , as confirmed by gas loaded structural analysis.<sup>83</sup> This unique design endows Al-PyDC with the capacity to effectively purify  $C_2H_4$  from mixtures containing  $C_2H_2$  and  $C_2H_6$  in one step. Furthermore, this MOF has been shown to be highly stable under harsh conditions (Fig. 3c).

By expanding the dimensions of the individual organic sites to nitrogen-containing heterocyclic ligands, including indole-5-carboxylic acid (ind), benzimidazole-5-carboxylic acid (bzz), and indazole-5-carboxylic acid (izo), the functionalized MOF-808 materials attain a uniform distribution and specific configuration of such extensive binding sites. MOF-808-Bzz has multiple strong supramolecular interactions, providing an excellent geometric configuration for  $C_2H_6$ , resulting in the highest  $C_2H_2/C_2H_4$  (1/99, v/v) and  $C_2H_6/C_2H_4$  (50 : 50, v/v) selectivity among these four materials (Fig. 3d).<sup>53</sup>

To achieve efficient  $C_2H_4$  and  $C_2H_6$  separation, an inverse interaction, namely the stronger host– $C_2H_6$  interaction, has been considered. Inspired from the structures of metalloenzymes,  $Fe_2(O_2)(dobdc)$  exhibited a preferential binding of  $C_2H_6$  over  $C_2H_4$ .<sup>46</sup> High-resolution neutron powder diffraction (NPD) measurements revealed that the peroxy site binds  $C_2D_6$  through a C–D/O interaction, of which the D···O distance varies in a narrow range of 2.17 to 2.22 Å. In addition, the nonplanar  $C_2D_6$  molecule matches better to the pore surface in  $Fe_2(O_2)(-dobdc)$  than the planar  $C_2D_4$  molecule, resulting in stronger hydrogen bonds with the Fe-peroxy site and stronger van der Waals interactions with the ligand surface (Fig. 3e).

In light of the terminal acidic H on  $C_3H_4$ , the basic organic site strategy has also been explored in the design of MOFs for the removal of trace  $C_3H_4$  from  $C_3H_6$ . Taking the pore-partitioned MOF as a platform, aminophthalic acid (NH<sub>2</sub>-BDC) was selected to construct the framework.<sup>84</sup> The amino sites, which are densely arranged, possess a distinctive capacity to recognise the acetylic and methyl hydrogen atoms of  $C_3H_4$ , resulting in the formation of short hydrogen bonds. Moreover, the customised nanospace induces  $C_3H_4\cdots C_3H_4$  molecules to aggregate together with short H···C distances. Although the NH<sub>2</sub> group also interacts with  $C_3H_6$ , the interaction distance is longer, yielding a sharp adsorption difference (84.5 cm<sup>3</sup> g<sup>-1</sup>) at about 1 kPa, 298 K. Furthermore, such an interaction difference was also finely illustrated by *in situ* infrared spectroscopy measurement.

**2.1.3 Multiple sites.** MOFs with multiple sites show a promising future in the field of gas separation, due to the systematically tuned chemical properties, the associated pore size changes and the excellent stability. Typically, the fluorine atom of the inorganic fluorinated anions, supported as the pillar of the MOF framework, has high electronegativity, high ionisation potential and low polarizability, which can drastically change the gas adsorption characteristics of the MOFs.<sup>85</sup>

To further promote the ability to selectively capture  $C_2H_2$  with a terminal acidic H, the weakly basic hybrid anion  $MF_6^{2-}$  (M = Si, Ti and Zr) was selected. As a typical example, SIFSIX-1-Cu [SIFSIX, hexafluorosilicate ( $SiF_6^{2-}$ ); 1, 4,4'-bipyridine]] was prepared.<sup>34</sup> The prototypical primitive cubic network has previously been reported for high volumetric  $CH_4$  and  $CO_2$  uptake.<sup>86,87</sup> Notably, the periodically arrayed  $SiF_6^{2-}$  allowed

preferential binding of  $C_2H_2$  molecules through strong C–H···F H-bonding (2.017 Å) and van der Waals (vdW) interactions with the organic linkers. Constrained by the narrow pores, the four neighbouring adsorbed molecules assemble to form a gas cluster through multiple  $H^{δ+}\cdots C^{δ-}$  dipole–dipole interactions, further increasing the energy for adsorption. Comparably, the calculated H-bond distances between  $C_2H_4$  and  $SiF_6^{2-}$  sites are longer than those between  $C_2H_2$  and  $SiF_6^{2-}$  sites. This unique interaction allows SIFSIX-1-Cu to show balanced performance between  $C_2H_2$  uptake and  $C_2H_2/C_2H_4$  selectivity (Fig. 4a). This idea has been further expanded in the family of cubic net of SIFSIX-1-Cu, SIFSIX-2-Cu (2: 4,4'-dipyridylacetylene), SIFSIX-2-Cu-i, SIFSIX-3-Cu (3: pyrazine), SIFSIX-3-Zn, SIFSIX-3-Ni, and TIFSIX-2-Cu-i (TIFSIX =  $TiF_6^{2-}$ ) by changing the length of the linkers, the node, and/or the framework interpenetration for separation of  $C_2H_4$  from ternary ( $C_2H_2/C_2H_6/C_2H_4$ ) or quaternary ( $CO_2/C_2H_2/C_2H_6/C_2H_4$ ) mixtures (Fig. 4a).<sup>88</sup>

By changing pyridine to imidazole at the coordination sites of the linear ligand, a new MOF (NTU-67) with a trap-and-flow

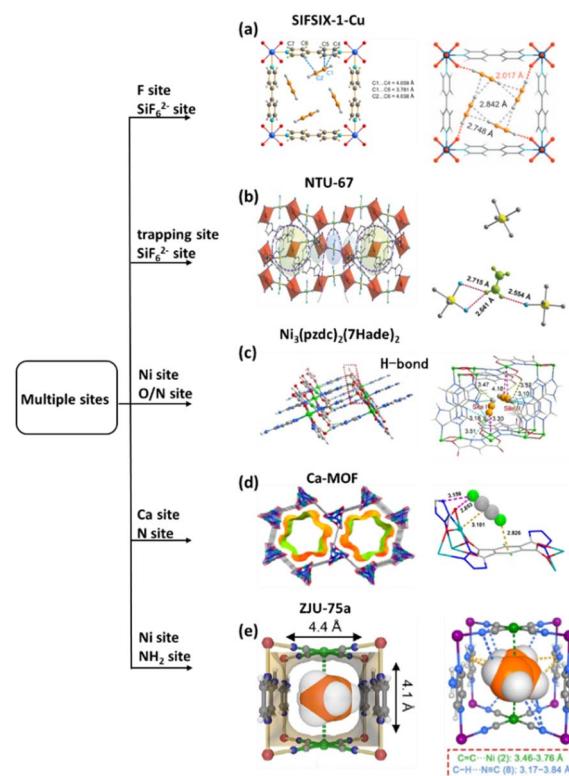


Fig. 4 (a) Optimized  $C_2H_2$  configuration and adsorption binding sites in SIFSIX-1-Cu. (b) View of the inner space in channel A, and view of the structure of a molecular trap with pure gas of NTU-67 $\supset C_2H_4$ . (c) Coordination environment of  $Ni_3(pzdc)_2(7Hade)_2$ , binding configurations of  $C_2H_2$ , and the open  $Ni^{2+}$  strong polarization capacity and the calculated  $C\equiv C-H$  bond angle. (d) Coordination environment of  $Ca^{2+}$  ions and preferential adsorption sites for  $C_2H_2$ . (e) Densely arranged high-density strong binding sites and  $C_3H_6$  adsorption sites based on theoretical calculations. Reprinted with permission from ref. 34. Copyright 2016 The American Association for the Advancement of Science, ref. 36. Copyright 2022 John Wiley & Sons, ref. 85. Copyright 2020 John Wiley & Sons, ref. 89. Copyright 2021 American Chemical Society, ref. 90. Copyright 2023 John Wiley & Sons, respectively.



channel structure was prepared by crystal conversion. With the narrowest window aperture of 3.4 Å, the flow channel acted as a sieve channel for faster diffusion of the linear molecules of  $\text{C}_2\text{H}_2/\text{CO}_2$  (3.3 Å) and the butterfly-like molecule of  $\text{C}_2\text{H}_4$  (4.2 Å). Meanwhile, the confined nanospace regulated by three  $\text{SiF}_6^{2-}$  anions and six imidazoles in another channel works as a molecular trap for adsorbed molecules. Crystallographic study under pure gas revealed that the linear  $\text{C}_2\text{H}_2$  interacts with  $\text{SiF}_6^{2-}$  anions and imidazole carbon to form short hydrogen bonds in the range of 2.470–2.984 Å. In addition, the existing hydrogen bonds between the adsorbed  $\text{C}_2\text{H}_2$  molecules enable the formation of gas clusters, which benefits the  $\text{C}_2\text{H}_2$  uptake at very low pressure. For  $\text{CO}_2$ , they are observed in the molecular trap around the  $\text{SiF}_6^{2-}$  anions, forming a typical dipolar–dipolar ( $\text{Si}-\text{F}\cdots\text{C}=\text{O}: \delta^+ - \delta^- \cdots \delta^+ = \delta^-$ ) interaction. However, relatively longer hydrogen bonds of  $\text{F}\cdots\text{H}-\text{C}_{\text{C}_2\text{H}_4}$  (2.554–2.715 Å) were observed for  $\text{C}_2\text{H}_4$ . Notably, the competitive binding of the gases was validated by crystallographic study under mixed gas. Therefore, NTU-67 was able to directly harvest highly-pure  $\text{C}_2\text{H}_4$  from the ternary ( $\text{CO}_2/\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$ ) mixtures (Fig. 4b).<sup>36</sup>

To investigate the combined effect of open metal sites and electronegative sites in a single domain, a super-microporous MOF,  $\text{Ni}_3(\text{pzdc})_2(7\text{Hade})_2$  was designed and prepared. This structure features S-shaped one-dimensional ultramicroporous channels decorated with a high-density of open Ni sites and electronegative oxygen and nitrogen sites. First-principles density functional theory (DFT) calculations revealed that  $\text{C}_2\text{H}_2$  interacts with the Ni site. The slight bending of the  $\text{C}\equiv\text{C}-\text{H}$  bond angle suggests that the unsaturated  $\text{Ni}^{2+}$  site can polarize the  $\text{C}_2\text{H}_2$  molecule. In addition, the highly polar  $\text{H}_{\text{C}_2\text{H}_2}$  interacts with the basic or electronegative O and N sites on the pore surface through electrostatic interactions at very short distances of 3.10–3.60 Å. In contrast,  $\pi$ -complexation occurs between Ni and  $\text{C}_2\text{H}_4$  with a longer distance of 4.00–4.40 Å, despite the vdW interaction between  $\text{C}_2\text{H}_4$  and the pore surface being very similar in type and geometry to that observed for  $\text{C}_2\text{H}_2$ . Therefore, the difference in the synergistic effect significantly improves the adsorption capacity and selectivity of  $\text{C}_2\text{H}_2$  from  $\text{C}_2\text{H}_4$  containing mixtures (Fig. 4c).<sup>85</sup>

Another example is Ca-MOF, derived from a N, O-donor ligand 2,5-di(2*H*-tetrazol-5-yl) pyrophosphate.<sup>89</sup> This material exhibits unique metal carboxylate-nitrogen heterocyclic oxide units, and a high density of open metal sites and organic functional sites that enable it to selectively adsorb and separate  $\text{C}_2\text{H}_2$ . This interaction of the open metal sites with the  $\pi$ -electron of the  $\text{C}_2\text{H}_2$  molecules is a key feature, while the organic groups play a regulatory role in the pore structure and surface properties through organic sites. This group forms hydrogen bonds and other interactions with  $\text{C}_2\text{H}_2$ , thereby generating additional binding forces. Monte Carlo simulations revealed that  $\text{C}_2\text{H}_2$  forms strong  $\text{C}-\text{H}\cdots\text{O}/\text{N}$  bonds with benzene rings and interacts with  $\text{Ca}^{2+}$  sites through  $\text{M}\cdots\pi$  interactions. In comparison,  $\text{C}_2\text{H}_4$  lacks  $\text{C}-\text{H}\cdots\pi$  interaction (Fig. 4d). Additionally, CuTiF<sub>6</sub>-TPPY (TPPY: 5,10,15,20-tetra(4-pyridyl)-21*h*,23*h*-porphyrin) with semi-cage-like 1D channels decorated with synergistic binding sites of  $\text{TiF}_6^{2-}$  and TPPY exhibits a noticeable adsorption of  $\text{C}_2\text{H}_2$  and  $\text{C}_2\text{H}_6$  over  $\text{C}_2\text{H}_4$ .<sup>67</sup>

Therefore, efficient single-step  $\text{C}_2\text{H}_4$  purification from a  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  mixture has been achieved based on this MOF. Similar to the structure of  $\text{Fe}(\text{pyrazine})\text{Pt}(\text{CN})_4$ , a new MOF  $\text{Co}(\text{pyz-NH}_2)[\text{Ni}(\text{CN})_4]$  (ZJU-75a, pyz-NH<sub>2</sub> = 2-aminopyrazine) has been prepared for the separation of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$ .<sup>90</sup> The open Ni site and the organic NH<sub>2</sub> sites point into the narrow channel (4.4 Å), allowing multiple interactions with  $\text{C}_3\text{H}_6$ , including double  $\pi$ -complexation interactions (3.46 and 3.76 Å) with face-to-face packed Ni sites and multiple supramolecular interactions with N sites ( $\text{C}-\text{H}\cdots\text{N}_{\text{CN}}$ : 3.17–3.84 Å and  $\text{C}-\text{H}\cdots\text{N}_{\text{NH}_2}$ : 2.79–3.24 Å). In contrast, only weak supramolecular interactions ( $\text{C}-\text{H}\cdots\text{N}_{\text{CN}}$ : 3.07–3.81 Å) and  $\text{C}-\text{H}\cdots\text{N}_{\text{NH}_2}$ : 2.63–3.55 Å) were observed in host@ $\text{C}_3\text{H}_8$ , due to the lack of a double bond in  $\text{C}_3\text{H}_8$  (Fig. 4e).

**2.1.4 Molecular sieving.** Based on the molecular size or shape, molecular sieve separation that allows smaller molecules to pass, but not large ones, has been considered as an effective strategy for adsorption separation of light hydrocarbons. Complete separation ensures the purity of the adsorbed gas and enables infinite selectivity. However, the molecular sizes and shapes of light hydrocarbons are very similar, making it challenging to construct MOFs with open pores that are located between the molecular diameters of the gases, particularly those with the same number of carbon atoms.

As a typical example,  $[\text{Ca}(\text{C}_4\text{O}_4)(\text{H}_2\text{O})]$  (UTSA-280), synthesized from calcium nitrate and squaric acid, has two parallel 1D open cylindrical channels with similar cross-sectional areas of about 14.4 Å<sup>2</sup>.<sup>70</sup> Importantly, this value is larger than the minimum cross-sectional area of  $\text{C}_2\text{H}_4$  (13.7 Å<sup>2</sup>), but smaller than that of  $\text{C}_2\text{H}_6$  (15.5 Å<sup>2</sup>). Therefore, it exhibits exclusive  $\text{C}_2\text{H}_4$  adsorption, but not of  $\text{C}_2\text{H}_6$  at 298 K. This observation was further confirmed by the adsorption isotherms of the two gases at 273 and 195 K. Further diffraction experiments and calculations show that the  $\text{C}_2\text{H}_4$  molecules adopt a head-to-head configuration inside the 1D channel, associated with weak  $\text{C}-\text{H}\cdots\text{O}$  hydrogen bonding,  $\pi\cdots\pi$  stacking and vdW interactions with the rings of the ligand or coordinated water molecules. However,  $\text{C}_2\text{H}_6$ , regardless of its orientation, is strongly restricted by the aperture of the channel (Fig. 5a).

Moving to the target of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures, the two have <0.4 Å size difference, and design and construction of the sieving channel becomes more challenging. Selecting or tailoring the length of the ligands is not effective, as the required change is located in a wider range. Inspired by the sub-Å level size change of the hybrid ions, a chemically stable fluorinated MOF, KAUST-7, was prepared according to the reticular chemistry approach.<sup>32</sup> Initially, by changing the  $\text{SiF}_6^{2-}$  pillar in SIFSIX-3-Ni (adsorbs both  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  due to the free rotation of the pyrazine ligands) to a slightly bigger cation  $\text{NbOF}_5^{2-}$ , the short proximal distance between adjacent F atoms provides a plausible window opening of 3.0–4.8 Å, associated with the restricted rotation and tilting of the pyrazine linker. Therefore, KAUST-7 displayed full exclusion of  $\text{C}_3\text{H}_8$  from  $\text{C}_3\text{H}_6$ -containing mixtures. Sieving separation of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  has also been observed in Y-abtc (abtc = 3,3',5,5'-azobzenene tricarboxylates), which has cage-like pores connected through small windows of 4.72 Å (Fig. 5b).



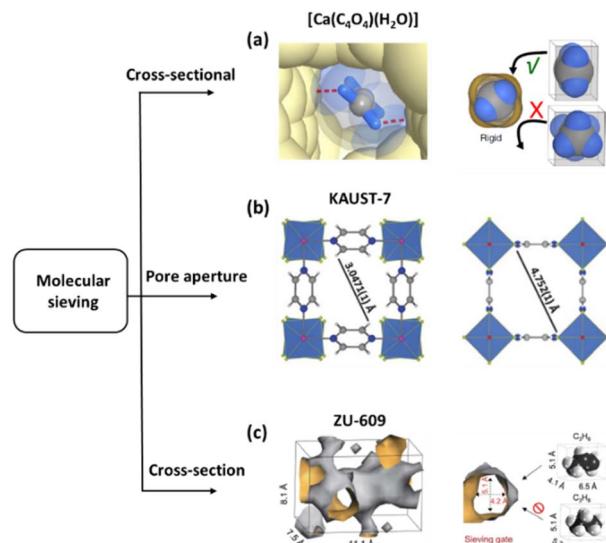


Fig. 5 (a) Preferential binding site for  $\text{C}_2\text{H}_4$  molecules, and size/shape sieving based on the minimum cross-sectional areas of  $\text{C}_2\text{H}_4$  and  $\text{C}_3\text{H}_6$  molecules. (b) Crystal structure of NbOFFIVE-1-Ni, and simulation of the maximum open structure of NbOFFIVE-1-Ni. (c) Sieving gate of ZU-609 and the molecular sizes of propylene and propane. Reprinted with permission from ref. 70. Copyright 2018 Springer Nature Limited, ref. 32. Copyright 2016 The American Association for the Advancement of Science, ref. 33. Copyright 2023 The American Association for the Advancement of Science, respectively.

Although the strategy of molecular sieving can provide complete separation of the mixtures, the narrow pore strongly restricts molecule diffusion, a long-standing issue of adsorption separation. ZU-609, a 2D network composed of an inorganic metal node and organic linkers (EDS<sup>2-</sup>, 1,2-ethanedisulfonate; dps, 4,4'-dipyridyl sulfide), exhibits a large 1D channel with a size of 7.5–11.1 Å.<sup>33</sup> Meanwhile, such a channel was connected by a narrow neck (4.2 × 5.1 Å). This cross-sectional area falls just between the molecular dimensions of  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$ . In other words, by incorporating a molecular sieve gate and a fast diffusion channel in a single domain, ZU-609 exhibits precise exclusion of  $\text{C}_3\text{H}_8$  from  $\text{C}_3\text{H}_6$ -containing mixtures, as well as fast  $\text{C}_3\text{H}_6$  adsorption kinetics. Particularly, the diffusion coefficient of this MOF is 1–2 orders of magnitude higher than that of KAUST-7 and Co-gallate. The rapid diffusion and high sorption were finely supported by *in situ* PXRD analysis and dispersion corrected calculation (Fig. 5c).

**2.1.5 Diffusion controlled.** The strategy of molecular sieving with accelerated adsorption–diffusion undoubtedly has significant advantages in adsorption science; however, precise design and avoidance of sieving losses, given the flexible nature of the target MOF, are major challenges. Synergistic thermodynamic and kinetic adsorption, mainly determined by strong binding sites and pore size/shape, holds unique advantages. For instance, based on the bis(5-methyl-1*H*-1,2,4-triazol-3-yl) methane ligand (btm),  $[\text{Zn}_2(\text{btm})_2]$  (MAF-23) was prepared. However, the methylene group in the framework cannot distinguish the C2/C3 hydrocarbons.<sup>88</sup> After heating this MOF under oxygen flow, almost half of the btm ligands were oxidized, forming a new ligand bis(5-methyl-1,2,4-triazol-3-yl) methanone

(H<sub>2</sub>btm). Due to the similar properties between the methylene group and carbonyl group, the channel size/shape of MAF-23-O is basically the same as that of the precursor framework. MAF-23-O demonstrated faster adsorption of  $\text{C}_3\text{H}_6$  than that of  $\text{C}_3\text{H}_8$ ; this is because the lone pair of the carbonyl oxygen interacts with the  $\text{sp}^2$  C–H bond of  $\text{C}_3\text{H}_6$ , but not for  $\text{C}_3\text{H}_8$  (Fig. 6a). In addition, this work also highlights the importance of post-synthetic modification, a reliable strategy for achieving the desired structure–property relationship compared to direct synthesis from the corresponding building blocks.

To further tune the synergistic effect of thermodynamic and kinetic adsorption performance of  $\text{C}_3\text{H}_6$  from  $\text{C}_3\text{H}_8$ , a group of MOFs were synthesized using chiral ligands. With L-malic acid, the L-mal-MOF has a homogeneous one-dimensional pore structure measured to be 5.3 × 5.5 Å. In contrast, the racemic material DL-mal-MOF synthesized from a mixture of D/L-malic acid ligands exhibited periodic contraction–expansion pore structures, forming quasi-discrete pore structures. This difference was caused by the different orientations of the chiral group in the nanospace. Similar to MAF-23-O, abundant oxygen atoms were found in these contraction–expansion pore structures. Adsorption isotherms showed that the DL-mal-MOF has a significantly higher equilibrium-kinetic combined selectivity of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$ , which was further illustrated by energy fluctuations and corresponding binding energy during the diffusion of the  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  molecules in such quasi-discrete pore structures.<sup>27</sup> This result suggests that the narrow-neck channel, along with the O-rich environment, may make the MOF a  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  separator (Fig. 6b). Utilizing a similar concept, a phosphate anion ( $\text{PO}_4^{3-}$ ) functionalized MOF ZnAtzPO<sub>4</sub> with a narrow (4.94 Å)-neck (3.82 Å) channel, has been reported for  $\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  separation (Fig. 6c).<sup>35</sup>

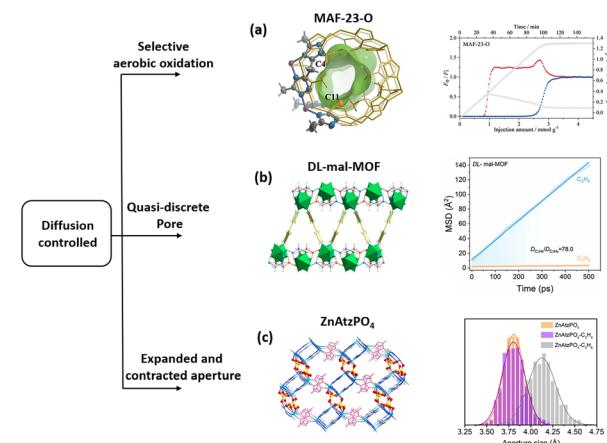


Fig. 6 (a) Crystal and pore structures of MAF-23-O, and breakthrough curves and adsorption kinetic curves using an equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixture at 298 K and 1 atm. (b) Structure of DL-mal-MOF and MD-derived self-diffusion rates of  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$ . (c) Three-dimensional structure of ZnAtzPO<sub>4</sub> and distribution of the aperture size of the bottleneck. Reprinted with permission from ref. 88. Copyright 2019 John Wiley & Sons, ref. 27. Copyright 2023 The American Chemical Society, ref. 35. Copyright 2020 Science, respectively.



## 2.2 Oriented design strategies in soft MOFs

Soft MOFs, a kind of crystalline porous solid with reversible phase transformation, have been considered as the most interesting discovery in materials chemistry in the last two decades. The combination of diverse structural nature and an external stimulus (such as gas adsorption, temperature, pressure, light, and electric field) creates emergent softness of the frameworks; particularly, some of them can show fully different structural responses to the gases, even if they have very similar properties. This unique characteristic thereafter sheds light on the development of adsorption based light hydrocarbon separation technology with significant vitality. According to the modified parts of the framework, they can be broadly divided into three types: global softness, local softness and confined softness. However, the design and construction of the desired softness for the accurate detection of light hydrocarbons, as well as the thorough understanding of the softness involved, remains a critical challenge.

**2.2.1 Global softness of the framework.** The global softness of MOFs refers to their ability to exhibit significant structural flexibility and adaptability, usually caused by layer sliding, framework expansion, bond formation and breaking, *etc.* Their ability to structurally adapt enhances their performance and opens up opportunities for the design of new MOFs tailored to specific industrial needs. With the discovery of the first soft MOF ZIF-7 (gate-opening mechanism) for  $C_2H_4$  and  $C_2H_6$  separation,<sup>38</sup> the design or tuning of structural softness, usually including a rapid change, step change and gradual change, for light hydrocarbon separation became a hot topic, as it is significant for exclusive molecular recognition.

**2.2.1.1 Rapid change.** Typically, an ultramicroporous MOF UTSA-300 ( $[\text{Zn}(\text{dps})_2(\text{SiF}_6)]$ , dps = 4,4'-dipyridylsulfide) with 2D channels of about 3.3 Å was reported. Interestingly, the fully activated framework transformed into a closed pore phase with 0D nanospaces. This is due to the conformational changes of the dps ligands and the rotation of the  $\text{SiF}_6^{2-}$  ions, leading to a shrinkage of the framework. However, this 0D nanospace starts to open under  $C_2H_2$  at 273 K, but not for  $C_2H_4$  and  $\text{CO}_2$ . Neutron powder diffraction and modelling studies pointed out that two F atoms from the two adjacent  $\text{SiF}_6^{2-}$  ions bind a  $C_2H_2$  molecule (*via* short hydrogen bonds) in a head-on configuration, leading to a recovery of the pore system from 0D to 2D.<sup>39</sup> The framework change allows UTSA-300a to selectively capture  $C_2H_2$  from  $C_2H_4$  or  $\text{CO}_2$  containing mixtures.

In contrast, NTU-65, the global soft framework demonstrated different structural changes, resulting in efficient  $C_2H_4$  purification ability from  $C_2H_2/\text{CO}_2/C_2H_4$ .<sup>92</sup> The as-synthesized NTU-65, constructed from 1,4-di(1*H*-imidazole-1-yl)benzene,  $\text{Cu}^{2+}$  and  $\text{SiF}_6^{2-}$ , exhibits a 3D framework with pcu topology. The two adjacent, rather than the two opposite, F atoms on  $\text{SiF}_6^{2-}$  join the coordination of the framework, while other F atoms form hydrogen bonds with the ligands. In addition, NTU-65 has two types of nanochannels with opening sizes of  $2.6 \times 3.4$  and  $5.2 \times 6.3$  Å<sup>2</sup>. Therefore, the activated framework showed sharply different PXRD and parameters of the unit cell, indicating the global framework change. Interestingly, this

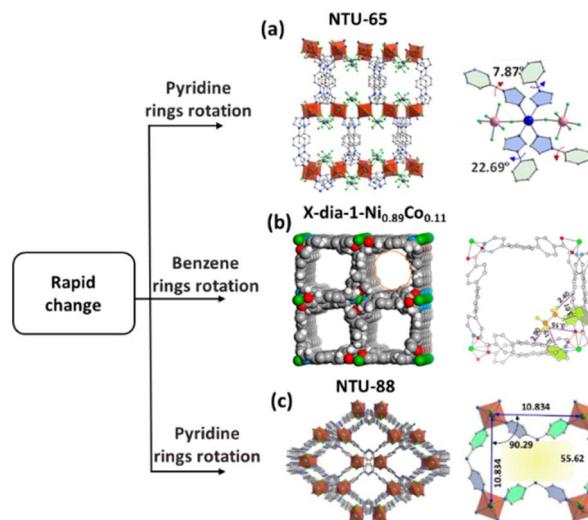


Fig. 7 (a) The crystal structure of NTU-65 *via* the  $c$  axis and benzene ring rotation of NTU-65. (b) The crystal structure of X-dia-1- $\text{Ni}_{0.89}\text{Co}_{0.11}$  and  $C_2\text{H}_4$  and  $C_2\text{H}_6$  binding sites at X-dia-1- $\text{Ni}_{0.89}\text{Co}_{0.11}$ . (c) The crystal structure of NTU-88 *via* an axis and open phase of NTU-88o under  $C_3\text{H}_4$  at 100 kPa. Reprinted with permission from ref. 92. Copyright 2020 John Wiley & Sons, ref. 56. Copyright 2024 American Chemical Society, ref. 69. Copyright 2023 John Wiley & Sons, respectively.

activated framework showed a temperature-dependent response for  $C_2H_2$ ,  $\text{CO}_2$ , and  $C_2H_4$  at 195 K, but only for  $C_2H_2$  at 298 K. On further temperature optimization to 263 K,  $C_2H_2$  and  $\text{CO}_2$  were adsorbed, resulting in a one-step purification of  $C_2H_4$  (Fig. 7a).

In addition, sudden pore opening of X-dia-1-Ni occurs on  $C_2H_6$ , but not on  $C_2H_4$ , affording direct harvesting of  $C_2H_4$  in the adsorption step.<sup>56</sup> The pressure of  $C_2H_6$  triggered sudden opening can be finely tuned through partial  $\text{Co}^{2+}$  doping, due to the better shape-fitting between the host and  $C_2H_6$  under increased pressure. Further modelling calculation results confirmed that the open phase of X-dia-1-Ni exhibited strong  $C_2H_6$  interaction through multiple  $\text{C}-\text{H}\cdots\text{O}$  bonding with distances of 2.40 and 3.61 Å, whereas only two  $\text{C}-\text{H}\cdots\text{O}$  bonds with distances of 3.75 and 3.83 Å were observed between the host and  $C_2H_4$  (Fig. 7b).

On the base of an NTU-65 precursor with Cu-F (from  $\text{SiF}_6^{2-}$ ) and Cu-N (from ligand) coordination bonds, the same group developed a new approach to fine regulate the gradient gate-opening in a series of MOFs (NTU-65-FeZr, NTU-65-FeTi, NTU-65-CoZr and NTU-65-CoTi) *via* node substitution.<sup>51</sup> Due to the systematically altered strength of the coordination bond, the sole structural response toward  $C_3\text{H}_4$  in NTU-65-FeZr gradually evolves into a sequential response to  $C_3\text{H}_4$  (1.6 kPa),  $C_3\text{H}_6$  (19.4 kPa), and  $C_3\text{H}_8$  (57.2 kPa) in NTU-65-CoTi at 273 K, as the incorporation of multiple nodes determines the energy barrier of the global softness of the framework. This unique phenomenon allows NTU-65-CoTi to show sieve separation of  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  in one step.

Notably, co-adsorption is a kind of common phenomenon once the inherent large pore has been opened by the molecule



with strong host–guest interactions, which then leads to issues with selectivity and separation efficiency. To tackle this problem, we designed a rotational shutter (a rotated pyridyl ring) in NTU-88, a soft MOF composed of 4,4'-dipyridylnitride and  $\text{NiCl}_2$ .<sup>69</sup> The 2D coordination framework connected by halogen bonds exhibits a zig-zag channel with a rhomboid pore of  $5.3 \times 8.8 \text{ \AA}$ . But, the structure of the fully activated framework exhibits significant changes, particularly the sharply tailored open size of the channel ( $1.6 \text{ \AA}$ ). However, this framework demonstrated quick  $\text{C}_3\text{H}_4$  uptake, but no  $\text{C}_3\text{H}_6$  uptake. Structural models under typical pressure and modelling calculations showed that a rotation of the pyridine rings in L was observed in the open phase of this MOF, creating an open pore with a maximum size of  $4.4 \text{ \AA}$ , very close to the molecular size of  $\text{C}_3\text{H}_4$  ( $4.4 \text{ \AA}$ ), but smaller than that of  $\text{C}_3\text{H}_6$  ( $5.4 \text{ \AA}$ ). Thus, this is a rare soft example that can comprehensively suppress co-adsorption (Fig. 7c).

**2.2.1.2 Stepwise change.** It is worth noting that the adsorption isotherms of these corresponding MOFs show a clear step in a slightly wider pressure range. However, the guest- or stimuli-triggered phase transition in soft MOFs passes through a closed phase, an intermediate phase and then an open phase. Such stepwise phase transitions represent a novel and responsive platform for molecular recognition, even for light hydrocarbons. As an early example, the MIL-53-series demonstrates stepwise adsorption isotherms of  $\text{CO}_2$  and alkanes.<sup>93,94</sup> A similar stepwise  $\text{N}_2$  adsorption isotherm was also found in a flexible framework  $\text{Co}(\text{BDP})$ .<sup>95,96</sup> However, the stepwise structural change dominated by coordination bonds usually requires high energy input (high gas pressure), which could not distinguish the small difference in the light hydrocarbons. Inspired by the stepwise isotherms of  $\text{CO}_2$  in a partially interpenetrated MOF, MFM-202, an interpenetrated dia framework, showed a stepwise  $\text{C}_2\text{H}_2$  uptake at 195 K.<sup>97</sup> However, the unique function of such a stepwise change was not investigated in detail.

As an early reported soft MOFs, ELM-11 adopts a 2D square-grid coordination framework. The *trans*-axial positions of the metal centres were occupied by two  $\text{BF}_4^-$  anions. Interestingly, the author found S-shaped adsorption isotherms of  $\text{CO}_2$ ,  $\text{N}_2$ ,  $\text{O}_2$ ,  $\text{Ar}$ , and  $\text{CH}_4$ , caused by stepwise layer expansion. Utilizing this characteristic, ELM-11 was found to have a stepwise  $\text{C}_2\text{H}_2$  uptake and trace  $\text{C}_2\text{H}_4$  adsorption, resulting in efficient separation performance, when the partial pressure of  $\text{C}_2\text{H}_2$  changed from 50 kPa to 10 kPa.<sup>98</sup>

To develop the function of the MOFs that involves stepwise opening,  $\text{Zn}_2(\text{bpdc})_2(\text{bpee})$ , ( $\text{bpdc} = 4,4'$ -biphenyldicarboxylate;  $\text{bpee} = 1,2$ -bipyriylethylene) was investigated.<sup>71</sup> The framework showed emerging stepwise adsorption isotherms of short alkanes at 298 K, as well as  $\text{C}_2$  hydrocarbons. Raman spectra of the  $\text{C}_2\text{H}_6$  loaded sample revealed that the methyl group of  $\text{C}_2\text{H}_6$  interacts with the uncoordinated  $\text{C}=\text{O}_{\text{coo}}$  on the bpdc ligand, causing a decrease in the dihedral angle ( $\Delta\phi = -2.0^\circ$ ) between the two rings of the bpdc ligand. Comparably, the dihedral angle increases ( $\Delta\phi = 1.5^\circ$ ) when  $\text{C}_2\text{H}_4$  was adsorbed, reflecting a competitive alternative binding site with similar binding strength to that of the H-bonding. In other words, a higher pressure of  $\text{C}_2\text{H}_4$  ( $\pi$  orbital plays a minimal role) is required to

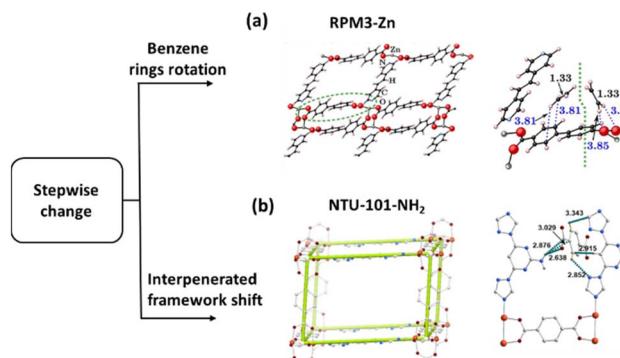


Fig. 8 (a) The lateral perspective of the RPM3-Zn structure and  $\text{C}_2\text{H}_2$ ,  $\text{C}_2\text{H}_4$  and  $\text{C}_2\text{H}_6$  binding sites at  $\text{Zn}_2(\text{bpdc})_2(\text{bpee})$ . (b) The crystal structure of NTU-101-NH<sub>2</sub>. Reprinted with permission from ref. 71. Copyright 2012 American Chemical Society, ref. 62. Copyright 2025 John Wiley & Sons, respectively.

open the gate of such a framework. Furthermore, the chain length of short alkanes is crucial for the gate opening of the pressure dependence ( $\text{C}_2 < \text{C}_3 < \text{C}_4$ ). Therefore, tuning the H-bond strength is shown to be a strategy for optimizing the MOFs with the desired gate-opening pressure (Fig. 8a).

Inspired by this finding and also by the dynamics of the interpenetrated MOF, we recently presented an approach to control the temperature-dependent dynamics in hydrogen-bonded interpenetrated frameworks. NTU-101-NH<sub>2</sub>, a single H-bond linked interpenetrated porous framework, exhibited stepwise structural dynamics in response to  $\text{CO}_2$ .<sup>62</sup> Importantly, this MOF showed gas and pressure dominated dynamics towards  $\text{C}_2\text{H}_6$  (37 kPa, 328 K) and showed an inverse ability for  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  separation at an elevated temperature of 328 K. This was due to the shift of the dynamics towards  $\text{C}_2\text{H}_6$  (37 kPa, 328 K) and  $\text{C}_2\text{H}_4$  (53 kPa, 328 K). This is because the displacement of the interpenetrating frameworks here requires a relatively weak stimulus, allowing the adsorption to be optimised in a higher temperature range. However, NTU-101, the precursor framework linked by three H-bonds, shows structural dynamics at very low temperatures. Therefore, hydrogen-bonded frameworks were expected to pave the way for the design of soft families capable of challenging separations at higher temperatures (Fig. 8b).

**2.2.1.3 Gradual change.** Sudden pore opening and stepwise pore opening both result from overcoming the clearly defined host–guest interactions, but the change is not always a sudden transition from a closed phase to an open phase. During the adsorption process, the pores or structure of MOF materials can be gradually adjusted to match the size and shape of the adsorbate molecules, resulting in a progressively enhanced host–guest interaction. This characteristic is different from the structural response of MOFs with a sudden global change and gradual global change.

Yet an earlier framework of CPL-1, one of the attractive adsorbents, has bridging pillar ligands and 2D porous sheets. The formed small channel with a pore size of  $4.0 \times 6.0 \text{ \AA}^2$  features abundant O atoms in a one-to-one fashion.<sup>99</sup>



Interestingly, the structure was shown to change gradually under increased pressure of  $C_2H_2$ , as the gradually enhanced host– $C_2H_2$  interaction promotes the changes in the structure.<sup>100</sup> The same group then investigated the separation potential of CPL-1 on  $C_2H_4$  and  $C_2H_6$ .<sup>101</sup> The flexibility of the framework allowed stepped adsorption isotherms, particularly for  $C_2H_4$  in the pressure range of 200 to 400 kPa, at 273 K; however, the  $C_2H_6$  uptake is very low in the measured region. Shifting the target to the C3 hydrocarbons, CPL-1 showed a gradually changing thermo-responsive gate-opening behavior towards  $C_3H_6$  in the temperature range of 273 to 288 K, but no structural response was found under  $C_3H_8$ .<sup>102</sup>

Due to the global framework change, the temperature assistant framework response usually occurred at relatively low temperature. To overcome this issue, a microporous MOF,  $[Zn_3OH]_2(btca)_2$  (JNU-1,  $H_2btca$  = benzotriazole-5-carboxylic acid), with gas-induced fit was synthesized. The activated JNU-1 exhibits a 1D microporous channel of 8 Å diameter with an accessible open zinc site.<sup>103</sup> A self-adaption framework change was observed upon increasing  $C_2H_2$  loading, namely a gradually closed form. Gas-loading crystal analysis and modelling calculations showed a side-on configuration of the trapped  $C_2H_2$  interacting with two adjacent open Zn sites. Furthermore, one  $C_2H_2$  molecule interacts with other two additional molecules bound to Zn sites *via* dipole interactions. The gradual increase in adsorption leads to enhanced host–guest interactions, which in turn trigger framework shrinkage. This phenomenon is distinctly different from the general observation of gradual pore enlargement of the framework associated with increased pressure of the adsorbed gas (Fig. 9a).

For the gradual change in the framework, we have recently found a new system that exhibits a subtle expansion and contraction of the channel neck. NTU-68 was prepared *via* crystal conversion of NTU-65 in boiling water.<sup>40</sup> Crystal analysis (from 200 to 340 K) revealed that the distance of the channel neck defined by F–F gradually expanded and then contracted to around 4 Å (turning point: 328 K), which falls between the centre dimensions of  $C_3H_6$  (3.44 Å) and  $C_3H_8$  (4.02 Å). Therefore, an unprecedented phenomenon of thermodynamically dominant  $C_3H_6$  uptake, but kinetically regulated  $C_3H_8$  adsorption behaviour, was observed. The calculated energy profile reveals that the maximum energy for passing through the cell is

42.8 kJ mol<sup>−1</sup>. However,  $C_3H_8$  has to overcome two strong interactions, corresponding to 46.3 and 63.8 kJ mol<sup>−1</sup>, respectively, as  $C_3H_8$  diffusion requires a configuration change. This temperature-dependent gradual change allows an enhancement in separation efficiency as the temperature increases from 273 to 298 K (Fig. 9b).

**2.2.2 Local softness of the framework.** Compared with the global softness of the MOF, the concept of local softness is defined as the partial movement of the MOF framework. Local softness, particularly in the context of a functional ligand, exhibits a unique advantage in its capacity for adaptive environmental adjustment, enabling precise recognition of guest molecules. This process results in a molecular sieving effect analogous to that of traditional MOFs. In other words, it is conceptually feasible to identify small differences in light hydrocarbon components by utilizing the local structural changes. The small changes in energy barriers offer prospects for expeditious material regeneration in the ensuing phase; nevertheless, they concomitantly engender challenges in precise structure design, and no commensurate theoretical framework has been formulated to date.<sup>104</sup> In the following section, a comparison is made between the local dynamics of the ligand (local node change usually causes global change) in the MOFs for light hydrocarbon separation and the more typical local dynamics of the ligand in other materials.

A coordination framework was synthesised and prepared from a butterfly-type ligand comprising isophthalic acid and phenothiazine-5,5-dioxide (OPTz) moieties (OPTz-IPA).<sup>72</sup> After activation, the framework features a nanocage connected by eight nanochannels. Interestingly, one OPTz moiety and one isophthalic unit adopt a face-to-face configuration, forming a gate of 3 Å inside the channel. Therefore, the thermal flipping of the OPTz units provides a gate function that enables different kinetic responses towards  $C_2H_4$  and  $C_2H_6$ , as well as  $O_2/Ar$ . It is worth noting that the ability for preferred adsorption is strongly temperature-dependent, of which the  $T_{max}$  is as low as 270 K for the adsorption separation of  $C_2H_4$  and  $C_2H_6$ . The function of such thermal local-dynamics was extended into the separation of water isotopologues, which have extremely similar physicochemical properties (Fig. 10a).<sup>105</sup>

Recently, a 3D porous MOF, named JNU-3, was prepared by using a branched ligand with carboxylate, triazole and pyridine coordination sites. JNU-3 possesses a 1D channel with dimensions of approximately 4.5 Å × 5.3 Å. The yielded cross-sectional area (23.85 Å<sup>2</sup>) is slightly larger than that of the minimum cross-sections of  $C_3H_6$  (19.34 Å<sup>2</sup>) and  $C_3H_8$  (18.17 Å<sup>2</sup>). In addition, a molecular pocket with an aperture size of 3.7 Å is lined up on both sides of the 1D channel. Given the larger kinetic parameters of  $C_3H_6$  and  $C_3H_8$ , it appears improbable that these pockets can accommodate them. However, the tilting and rotation of the “guard” moiety of aromatic rings on the ligand can enlarge the aperture (4.4 for  $C_3H_6$  and 4.7 Å for  $C_3H_8$ ) of the pocket, facilitating the adsorption of  $C_3H_6$  and  $C_3H_8$ . Consequently, the unique structure can be regarded as an orthogonal-array dynamic molecular sieve, allowing both fast adsorption–desorption kinetics and large capacity. The presence of a gas-responsive pocket has been demonstrated to facilitate the

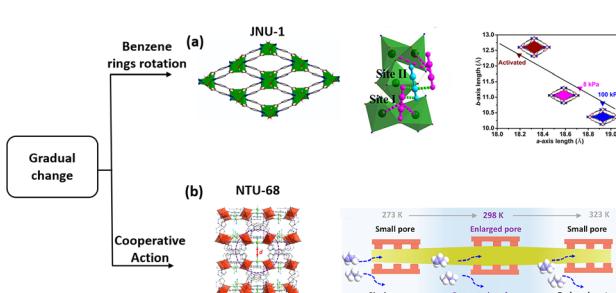


Fig. 9 The structure of JNU-1 (a) and illustration of the accelerated separation process in NTU-68a (b). Reprinted with permission from ref. 103. Copyright 2019 John Wiley & Sons, ref. 40. Copyright 2023 American Chemical Society, respectively.



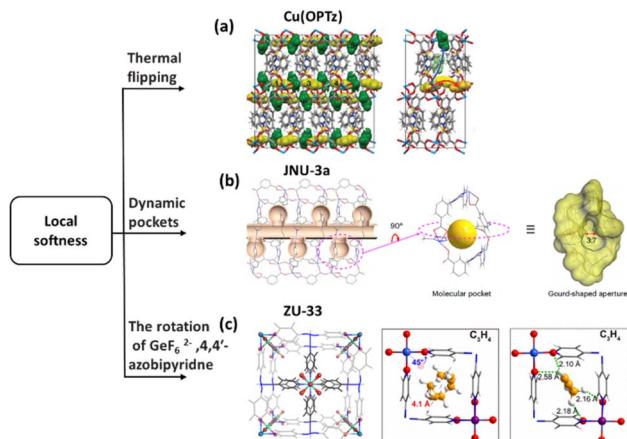


Fig. 10 (a) Simulation of the structure of activated Cu(OPTz)-loaded C<sub>2</sub>H<sub>4</sub> and the diffusion structure between the interlayer and intralayer. (b) Crystal structure of JNU-3a with molecular pockets and schematic diagram of molecular pockets. (c) The structure of ZU-33 and bonding sites for loaded C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>4</sub>(PD). Reprinted with permission from ref. 72. Copyright 2019 American Association for the Advancement of Science, ref. 106. Copyright 2024 Springer Nature Limited, ref. 107. Copyright 2022 Springer Nature Limited, respectively.

purification of C<sub>2</sub>H<sub>4</sub> from C<sub>2</sub>–C<sub>4</sub> mixtures in one-step (Fig. 10b).<sup>106</sup> In addition, this strategy has been expanded in HAF-1, which features channels and shrinkage throats, the latter being defined as narrower channels that connect the main channels and a molecular pocket.<sup>55</sup>

As a molecule separator, GeFSIX-14-Cu-i (ZU-33) can also exhibit an adaptable response to guest molecules.<sup>107</sup> The coordination of 4,4'-azopyridine with a GeF<sub>6</sub><sup>2-</sup> anion provides ZU-33 with local rotational flexibility, which plays a crucial role in the recognition of C<sub>2</sub>H<sub>2</sub>, C<sub>3</sub>H<sub>4</sub>, and C<sub>3</sub>H<sub>8</sub> (PD), and avoids the competitive adsorption of olefins and paraffins. The responsive properties to different stimuli endow this MOF with multiple regulations for challenging separations (Fig. 10c).

**2.2.3 Softness of the confined moiety in the framework.** Except the softness of the global framework and also the local moiety, we recently found a new kind of softness of the confined

moiety in a rigid framework.<sup>37</sup> The new MOF, (NTU-85) has a super-tetrahedral cluster (Cu<sub>10</sub>O<sub>13</sub>) and one-dimensional (1D) nanochannels consisting of lattice water molecules. Of particular significance is the observation that the lattice water molecules can be categorised into two distinct types: four molecules interact with the inner surface of the nanochannel *via* short hydrogen bonds, while a single molecule occupies the centre of the four water molecules and forms hydrogen bonds over a greater distance. In this regard, the central lattice water was accurately removed, resulting in the formation of a 1D water nanotube with an internal diameter of 4.5 Å within the rigid framework. The slight expansion and contraction of the water nanotube contributes exclusive C<sub>3</sub>H<sub>6</sub> adsorption, but not C<sub>3</sub>H<sub>8</sub> adsorption. This finding opens the route for establishing confined chemistry in MOFs (Fig. 11).

### 3. Applications

In recent years, there has been significant research interest in the field of adsorption separation of light hydrocarbons using MOFs. According to the demand for highly pure olefins, the separation process can be broadly classified into olefin-selective adsorption and alkane-selective adsorption.<sup>108–112</sup> To obtain pure olefins from an olefin-selective adsorption process, further desorption–adsorption cycles *via* inert gas sweeping or a vacuum pump are required. In contrast, alkane-selective adsorption has been shown to produce pure olefins directly, a process that is notably more energy efficient. This section will discuss recent research advances in the purification of C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub> from light hydrocarbon mixtures, particularly those with the same carbon numbers, in recent years.<sup>2,111,113</sup> Furthermore, we compared MOFs with framework dynamics induced by light hydrocarbons (Table 4), and the separation performance and the isosteric heat of adsorption ( $Q_{st}$ ) of MOFs used for light hydrocarbon separation (Table 5).

#### 3.1 C<sub>2</sub>H<sub>4</sub> harvesting from the desorption process

**3.1.1 C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> feed gas.** As a molecular sieving MOF, UTSA-280 demonstrated high C<sub>2</sub>H<sub>4</sub> capacity, up to 2.5 mmol g<sup>-1</sup>, at 298 K and 1 bar, whereas only a trace amount of C<sub>2</sub>H<sub>6</sub> (0.098 mmol g<sup>-1</sup>) was adsorbed.<sup>70</sup> Therefore, the calculated C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> selectivity is very high. In the breakthrough experiments, the packed samples exhibited full capture of C<sub>2</sub>H<sub>4</sub> from equimolar C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> mixtures, while highly pure C<sub>2</sub>H<sub>6</sub> was released initially. Enriched C<sub>2</sub>H<sub>4</sub> can be recovered with high purity during the regeneration step, yielding a productivity of C<sub>2</sub>H<sub>4</sub> as high as 1.86 mol kg<sup>-1</sup>. Notably, the sieving effect of C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> remains unaffected by the presence of additional components in the feed gas. Nevertheless, the role of coordinated water in preserving the structural integrity and enhancing the separation performance was found to be crucial (Fig. 12a).

ZU-901 exhibits a unique 'S' shaped C<sub>2</sub>H<sub>4</sub> adsorption curve, providing a high C<sub>2</sub>H<sub>4</sub> working capacity of up to 1.36 mmol g<sup>-1</sup>, at 273 K and 1 bar.<sup>116</sup> Interestingly, the material showed a C<sub>2</sub>H<sub>4</sub> uptake of 0.19 mmol g<sup>-1</sup> at 0.1 bar and 273 K, followed by a rapid increase to 1.55 mmol g<sup>-1</sup> at 1 bar, signifying the C<sub>2</sub>H<sub>4</sub>

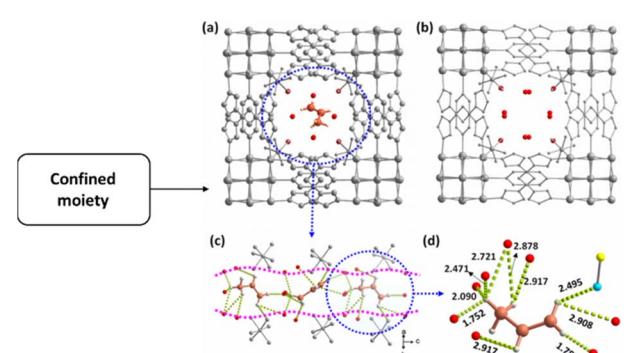


Fig. 11 Crystal structure of loaded C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> samples: (a) NTU-85-WNT-C<sub>3</sub>H<sub>6</sub> and (b) NTU-85-WNT-C<sub>3</sub>H<sub>8</sub> and (c, d) detailed H-bonding interactions of C<sub>3</sub>H<sub>6</sub> in one-dimensional nanochannels. Reprinted with permission from ref. 37. Copyright 2023 American Chemical Society.



Table 4 Light hydrocarbon-induced framework dynamics of the representative MOFs

MOFs	Reasons for phase changing	$P_{\text{gas}}$ for phase changing (kPa)	Temperature (K)	Ref.
ZIF-7	Configuration change in benzimidazole	$\text{C}_2\text{H}_6$ : 0.0008 $\text{C}_2\text{H}_4$ : 0.0012	298	38
NTU-88	Pyridine ring rotation	$\text{C}_3\text{H}_4$ : 1.7 $\text{C}_3\text{H}_6$ : no change	273	69
GeFSIX-dps-Zn	Pyridine ring rotation	$\text{C}_3\text{H}_4$ : 0.13	298	114
GeFSIX-dps-Cu	Pyridine ring rotation	$\text{C}_3\text{H}_4$ : 0.015		
NTU-65-CoTi	Benzene ring rotation	$\text{C}_3\text{H}_4$ : 1.6 $\text{C}_3\text{H}_6$ : 19.4 $\text{C}_3\text{H}_8$ : 57.2	273	51
$\text{Zn}_2(\text{bpdc})_2(\text{bpee})$	bpdc benzene ring rotation	$\text{C}_2\text{H}_2$ : 0.2 $\text{C}_2\text{H}_4$ : 0.4 $\text{C}_2\text{H}_6$ : 0.26	298	71
MFM-202	Benzene ring rotation	$\text{C}_2\text{H}_2$ : 2.7 $\text{C}_2\text{H}_4$ : 4.0 $\text{C}_2\text{H}_6$ : no change	195	97
NTU-101-NH <sub>2</sub>	Shifting of the interpenetration framework	$\text{C}_2\text{H}_6$ : 36 $\text{C}_2\text{H}_4$ : 57	293	62
X-dia-1-Ni	Pyridine ring and pyridine ring rotation	$\text{C}_2\text{H}_6$ : 51.7	273	56
ZU-13	Pyridine ring rotation	$\text{C}_3\text{H}_4$ : 0.05	298	115
NTU-68	Benzene ring rotation	Gradual change		40
Cu(OPTz)	Phenothiazine-5,5-dioxide rotation	Gradual change		72
TYUT-17	3-Methylisonicotinic rotation	$\text{C}_2\text{H}_6$ : <0.1	298	66
JNU-3	Pyridine ring rotation	Gradual change		65
ZU-33	4,4'-azopyridine ligand and $\text{GeF}_6^{2-}$ anion rotation	$\text{CH}_2=\text{C}=\text{CH}_2$ : 0.002	298	107
NTU-85	Dynamic of water nanotubes	Gradual change	298	37

working capacity of 1.36 mmol g<sup>-1</sup>. A similar adsorption phenomenon was observed upon increasing the temperature and pressure, to 298 K and 3 bar, respectively. Comparably, the C<sub>2</sub>H<sub>6</sub> capacity of ZU-901 is as low as 0.26 mmol g<sup>-1</sup> under both conditions. In breakthrough experiments, the clear separation interval confirmed the ability for C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> separation. Thanks to the relatively low binding energy, adsorbed C<sub>2</sub>H<sub>4</sub> can be rapidly regenerated and collected with >95% purity. Aspen adsorption simulation revealed that 99.51% purity of C<sub>2</sub>H<sub>4</sub> could be obtained with 75% recovery from a two-bed pressure swing adsorption process. In addition, this material can be prepared *via* stirred, associated with mother liquor circulation (Fig. 12b). Meanwhile, some other MOFs, such as Co(aip)(pyz)<sub>0.5</sub> (Fig. 12c)<sup>117</sup> and MOF-808-Bzz,<sup>53</sup> have been observed to demonstrate selective capture of C<sub>2</sub>H<sub>6</sub> from C<sub>2</sub>H<sub>4</sub>-containing mixtures. Furthermore, we have summarized the top ten C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> adsorption ratios, as illustrated in Fig. 16a. Notably, MAF-42 achieves an impressive adsorption ratio, up to 301 at 1 bar.<sup>118</sup> This summary serves as a valuable reference for efficient light hydrocarbon separation techniques.

### 3.2 C<sub>2</sub>H<sub>4</sub> harvesting from the adsorption process

**3.2.1 C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> feed gas.** Utilizing the interlayer and intralayer space, ZUL-100 and ZUL-101, which were synthesized from NbOF<sub>5</sub><sup>2-</sup> and three different organic ligands, showed highly selective capture of C<sub>2</sub>H<sub>2</sub> (selectivity: 175) from a C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> mixture (1/99, v/v).<sup>119</sup> In the breakthrough experiments, both ZUL-100 and ZUL-101 have excellent separation

performance. Notably, under humid conditions, they have been shown to achieve high-performance separation of trace amounts of C<sub>2</sub>H<sub>2</sub>, and to directly obtain polymer-grade C<sub>2</sub>H<sub>4</sub> with a yield of 121.2 mmol g<sup>-1</sup> and 103.6 mmol g<sup>-1</sup>, respectively (Fig. 13a).

Cu(OH)INA exhibited remarkable C<sub>2</sub>H<sub>2</sub> adsorption capacity performance within the temperature range of 272–313 K, even at lower pressures.<sup>120</sup> At 0.01 bar and 298 K, the adsorption amount of C<sub>2</sub>H<sub>2</sub> reached 44.01 cm<sup>3</sup> cm<sup>-3</sup>, while the adsorption amount of C<sub>2</sub>H<sub>4</sub> was found to be considerably lower. The adsorptive selectivity was calculated to be 105–41 and 71–35 at two distinct temperatures (298 K and 313 K), which confirmed the potential of the material for C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> (1/99, v/v) separation. Breakthrough experiments further validated the material's separation ability by obtaining high purity C<sub>2</sub>H<sub>4</sub> (>99.99%) with a productivity of 167 mL cm<sup>-3</sup>. Moreover, after the increase in flow rate, the separation efficiency (165 mL cm<sup>-3</sup>) remained close to that of the low flow rate. Furthermore, the large-scale synthesis, at least, can be scaled up to 8 L under stirring and 1.31 kg of product (91% yield) can be obtained (Fig. 13b).

Despite the fact that many of the MOFs have been the focus of research in regard to their potential for selective C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> capture, only a limited number of them have been shown to achieve a balance between C<sub>2</sub>H<sub>2</sub> uptake and C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> selectivity. SIFSIX-2-Cu-i with periodic SiF<sub>6</sub><sup>2-</sup> ions demonstrated a high C<sub>2</sub>H<sub>2</sub> adsorption capacity (2.1 mmol g<sup>-1</sup>, 0.05 bar, 298 K) and selectivity (39.7 to 44.8).<sup>34</sup> In contrast, SIFSIX-1-Cu exhibited exceptionally high C<sub>2</sub>H<sub>2</sub> uptake (8.5 mmol g<sup>-1</sup>) at 298 K and 1.0 bar, yet the superior C<sub>2</sub>H<sub>4</sub> uptake resulted in moderately high



Table 5 Typical MOFs for separation of light hydrocarbons

MOFs	Feed gas	Velocity of feed gas (mL min <sup>-1</sup> ) and working temperature (K)	Q <sub>st</sub> (kJ mol <sup>-1</sup> )	Purity and productivity	Ref.
UTSA-280	C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub> (1/1, v/v)	2.0, 298	C <sub>2</sub> H <sub>4</sub> ; 34.1	Highly pure C <sub>2</sub> H <sub>4</sub> , 1.86 mol kg <sup>-1</sup>	70
ZU-901	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	1.7, 273	C <sub>2</sub> H <sub>4</sub> ; 24.85	C <sub>2</sub> H <sub>4</sub> , 95%, —	123
Co(aip)[pyz] <sub>0.5</sub>	C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub> (1:1, v/v)	1.0, 298	C <sub>2</sub> H <sub>4</sub> ; 33.6	C <sub>2</sub> H <sub>4</sub> > 97%, 19.1 L kg <sup>-1</sup>	117
Fe <sub>2</sub> (O <sub>2</sub> ) <sub>2</sub> (dobdc)	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	5.0, 298	C <sub>2</sub> H <sub>6</sub> ; 66.8	C <sub>2</sub> H <sub>4</sub> , 99.99%, 0.79 mmol g <sup>-1</sup>	46
Ni-MOF 2	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	2.0, 298	C <sub>2</sub> H <sub>6</sub> ; 23.6	C <sub>2</sub> H <sub>4</sub> > 99.95%, 12 L kg <sup>-1</sup>	68
TYUT-17	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/9, v/v)	2.0, 298	C <sub>2</sub> H <sub>4</sub> ; 21.4	C <sub>2</sub> H <sub>4</sub> > 99.99%, 77.4 L kg <sup>-1</sup>	66
ZUL-100	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> (1/99, v/v)	1.25, 298	C <sub>2</sub> H <sub>2</sub> ; 65.3	C <sub>2</sub> H <sub>4</sub> > 99.9999%, 121.2 mmol g <sup>-1</sup>	119
Cu(OH)INA	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> (1/99, v/v)	2.0, 298	C <sub>2</sub> H <sub>4</sub> ; <40	C <sub>2</sub> H <sub>2</sub> ; 36.1	120
MOF-303	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub> (1/9/90, v/v/v)	1.25, 296	C <sub>2</sub> H <sub>4</sub> ; 29.6	C <sub>2</sub> H <sub>4</sub> > 99.99%, 167 mL cm <sup>-3</sup>	123
Al-PyDC	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> (1/99, v/v/v)	1.25, 296	C <sub>2</sub> H <sub>4</sub> ; 24.3	C <sub>2</sub> H <sub>4</sub> > 99.95%, 1.35 mmol g <sup>-1</sup>	123
MOF-808-Bzz	C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> /C <sub>2</sub> H <sub>6</sub> (1/1/1, v/v/v)	1.0, 298	C <sub>2</sub> H <sub>4</sub> ; 25.1	C <sub>2</sub> H <sub>4</sub> > 99.999%, 7.93 mmol g <sup>-1</sup>	83
MAC-4	C <sub>2</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (2/10/25, v/v/v)	7.0, 298	C <sub>2</sub> H <sub>4</sub> ; 35.3	C <sub>2</sub> H <sub>4</sub> > 99.95%, —	53
MOF-1	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>2</sub> /C <sub>2</sub> H <sub>4</sub> (3/3/10, v/v/v)	7.0, 298	C <sub>2</sub> H <sub>4</sub> ; 27.8	C <sub>2</sub> H <sub>4</sub> ; 27.8	132
Ni-dcpp-bpy	C <sub>2</sub> H <sub>2</sub> /CH <sub>4</sub> (1/1, v/v)	6.0, -298	C <sub>2</sub> H <sub>4</sub> ; 30.1	C <sub>2</sub> H <sub>4</sub> ; 30.1	133
SNNU-33	C <sub>2</sub> H <sub>2</sub> /CH <sub>4</sub> (5/2, v/v)	5.0, 298	C <sub>2</sub> H <sub>4</sub> ; 32.36	C <sub>2</sub> H <sub>4</sub> ; 32.36	133
NU-57	C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub> (1/1, v/v)	3.0, 298	C <sub>2</sub> H <sub>4</sub> ; 26.43	C <sub>2</sub> H <sub>4</sub> ; 26.43	133
JNU-3a	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1, v/v)	1.0, 298	C <sub>2</sub> H <sub>4</sub> ; 29.87	C <sub>2</sub> H <sub>4</sub> ; 29.87	132
HIAM-301	C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>8</sub> (95/5, v/v)	5.0, 298	C <sub>2</sub> H <sub>4</sub> ; 22.7	C <sub>2</sub> H <sub>4</sub> ; 22.7	132
ZJU-75	C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>8</sub> (1/1, v/v)	2.0, 296	C <sub>2</sub> H <sub>4</sub> ; 17.1	C <sub>2</sub> H <sub>4</sub> ; 17.1	132
FDC-4	C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>8</sub> (1/1, v/v)	1.0, 298	C <sub>3</sub> H <sub>6</sub> ; 25.3	C <sub>3</sub> H <sub>6</sub> ; 25.3	133
GefSIX-dps-Cu	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/9, v/v)	5.0, 298	C <sub>2</sub> H <sub>6</sub> ; 31.8	C <sub>2</sub> H <sub>6</sub> ; 31.8	134
SqI-NbOFFIVE-bpe-Cu-AB	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/99, v/v)	4.0, 300	C <sub>2</sub> H <sub>6</sub> ; 28.7	C <sub>2</sub> H <sub>6</sub> ; 28.7	134
		2.0, 298	C <sub>2</sub> H <sub>6</sub> ; 28.7	C <sub>2</sub> H <sub>6</sub> ; 28.7	134
		1.0, 298	C <sub>2</sub> H <sub>6</sub> ; 34.6	C <sub>2</sub> H <sub>6</sub> ; 34.6	134
		5.0, 298	C <sub>3</sub> H <sub>6</sub> ; 38.9	C <sub>3</sub> H <sub>6</sub> ; 38.9	135
		2.0, 296	C <sub>3</sub> H <sub>6</sub> ; 27.0	C <sub>3</sub> H <sub>6</sub> ; 27.0	130
		1.0, 298	C <sub>3</sub> H <sub>6</sub> ; 33.1	C <sub>3</sub> H <sub>6</sub> ; 33.1	130
		5.0, 298	C <sub>3</sub> H <sub>6</sub> ; 65.9	C <sub>3</sub> H <sub>6</sub> ; 65.9	65
		2.0, 296	C <sub>3</sub> H <sub>6</sub> ; 35.0	C <sub>3</sub> H <sub>6</sub> ; 35.0	65
		4.0, 300	C <sub>2</sub> H <sub>2</sub> ; 56.3	C <sub>2</sub> H <sub>2</sub> ; 56.3	65
		2.0, 298	C <sub>3</sub> H <sub>4</sub> ; 69.0	C <sub>3</sub> H <sub>4</sub> ; 69.0	74
		1.0, 298	C <sub>3</sub> H <sub>6</sub> ; 53.0	C <sub>3</sub> H <sub>6</sub> ; 53.0	114
				C <sub>3</sub> H <sub>6</sub> > 99.9%, 118 mmol g <sup>-1</sup>	136

Table 5 (Contd.)

MOFs	Feed gas	Velocity of feed gas (mL min <sup>-1</sup> ) and working temperature (K)	Q <sub>ext</sub> (kJ mol <sup>-1</sup> )	Purity and productivity	Ref.
Zn-BPZ-TATB	C <sub>2</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (2/10/25, <sup>54</sup> v/v/v)	8.0, 298	C <sub>2</sub> H <sub>6</sub> : 23.1 C <sub>2</sub> H <sub>4</sub> : 18.3 C <sub>3</sub> H <sub>6</sub> : 28.1 C <sub>2</sub> H <sub>4</sub> : 25.8 C <sub>3</sub> H <sub>6</sub> : 33.3 C <sub>2</sub> H <sub>4</sub> : 30.9–29.5 C <sub>2</sub> H <sub>6</sub> : 32.8–30.7	C <sub>3</sub> H <sub>6</sub> 99.5%, 38.2 L kg <sup>-1</sup> C <sub>2</sub> H <sub>4</sub> 99.9%, 12.7 L kg <sup>-1</sup>	3
Zn-MOF	C <sub>3</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/9, v/v)	5.0, 298	C <sub>2</sub> H <sub>6</sub> : 28.1 C <sub>2</sub> H <sub>4</sub> : 25.8 C <sub>3</sub> H <sub>6</sub> : 33.3 C <sub>2</sub> H <sub>4</sub> : 30.9–29.5 C <sub>2</sub> H <sub>6</sub> : 32.8–30.7	C <sub>2</sub> H <sub>4</sub> (>99.9 %), 98.71 L kg <sup>-1</sup>	137
Tb-MOF-76(NH <sub>2</sub> )	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> /Ar (5/5/90, v/v/v)	7.0, 298	—	C <sub>2</sub> H <sub>4</sub> , >99.9 %, —	138
X-dia-1-Ni <sub>0.89</sub> Co <sub>0.11</sub> Y-abte NKMOf-1-M (M = Cu or Ni)	(C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> : 1/9, v/v) C <sub>3</sub> H <sub>8</sub> /C <sub>3</sub> H <sub>6</sub> : 5/95, v/v) C <sub>3</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> /C <sub>3</sub> H <sub>6</sub> (0.5/0.5/99, v/v/v)	10, 298 4.0, 298 2.0, 298	C <sub>2</sub> H <sub>6</sub> : 57.4 C <sub>3</sub> H <sub>4</sub> : 65.1 (Ni), 67.2 (Cu) CH <sub>2</sub> =C=CH <sub>2</sub> : 54.0 (Ni), 45.2 (Cu)	C <sub>2</sub> H <sub>4</sub> , >99.9 %, — C <sub>2</sub> H <sub>4</sub> , 99.5%, — C <sub>3</sub> H <sub>6</sub> 99.996%, —	56 6 52
Ca-based MOF	C <sub>3</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> /C <sub>3</sub> H <sub>6</sub> (0.5:0.5:99, v/v/v)	2.7, 298	C <sub>2</sub> H <sub>6</sub> : 38.0 (Ni), 37.2 (Cu) C <sub>3</sub> H <sub>4</sub> : 55.5 CH <sub>2</sub> =C=CH <sub>2</sub> : 51.7	C <sub>3</sub> H <sub>6</sub> 99.95%, —	139
ZNU-2	C <sub>3</sub> H <sub>4</sub> /C <sub>3</sub> H <sub>6</sub> (1/9 or 1/99, v/v)	2.0, 298	C <sub>3</sub> H <sub>4</sub> : 43.2 C <sub>3</sub> H <sub>6</sub> : 35.5 C <sub>3</sub> H <sub>4</sub> : 44	C <sub>3</sub> H <sub>6</sub> > 99.996%, 37.8 or 52.9 mol kg <sup>-1</sup>	140
NTU-88 NTU-101-NH <sub>2</sub> JNU-9-CH <sub>3</sub>	C <sub>3</sub> H <sub>4</sub> /C <sub>3</sub> H <sub>6</sub> (1/1, v/v) C <sub>3</sub> H <sub>4</sub> /C <sub>3</sub> H <sub>6</sub> (1/99, v/v) C <sub>3</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> /C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1/1/1, v/v/v/v)	2.0, 298 4.0, 298 2.0, 298	C <sub>3</sub> H <sub>4</sub> : 43.5 C <sub>3</sub> H <sub>4</sub> : 34.6 CH <sub>2</sub> =C=CH <sub>2</sub> : 31.9 C <sub>3</sub> H <sub>6</sub> : 27.4 C <sub>3</sub> H <sub>8</sub> : 31.1	C <sub>3</sub> H <sub>6</sub> > 99.95%, — C <sub>3</sub> H <sub>6</sub> , 99.95%, 15.7 mL g <sup>-1</sup> C <sub>3</sub> H <sub>6</sub> ≥ 99.99%, —	69 62 60
PCP-IPA	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1, v/v)	3.7, 298	C <sub>3</sub> H <sub>8</sub> : 50.94 C <sub>3</sub> H <sub>6</sub> : 43.36	C <sub>3</sub> H <sub>6</sub> (99.99%), 15.2 L kg <sup>-1</sup>	129
CuZrF <sub>6</sub> -TPA	C <sub>3</sub> H <sub>4</sub> /CH <sub>2</sub> =C=CH <sub>2</sub> (1/1, v/v)	0.8, 298	C <sub>3</sub> H <sub>4</sub> : 46.1 CH <sub>2</sub> =C=CH <sub>2</sub> : 37.1	CH <sub>2</sub> =C=CH <sub>2</sub> > 99.95%, 4.7 mol L <sup>-1</sup>	141
NbOFFIVE-1-Ni NTU-85-WNT NTU-68 D,L-mal-MOF	C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1, v/v) C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1, v/v) C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1:1, v/v) C <sub>3</sub> H <sub>6</sub> /C <sub>3</sub> H <sub>8</sub> (1/1 or 95/5 v/v)	4.0, 298 0.5, 298 1.0, 298 2.0, 298	C <sub>3</sub> H <sub>6</sub> : 57.4 C <sub>3</sub> H <sub>6</sub> : 49.9 C <sub>3</sub> H <sub>6</sub> : 42.8 C <sub>3</sub> H <sub>6</sub> : 64.4	C <sub>3</sub> H <sub>6</sub> ~ 0.6 mol kg <sup>-1</sup> C <sub>3</sub> H <sub>6</sub> > 99.8%, 1.6 mL mL <sup>-1</sup> C <sub>3</sub> H <sub>8</sub> > 99.95%, — Highly pure C <sub>3</sub> H <sub>6</sub> , 2.1 L kg <sup>-1</sup>	32 37 40 27

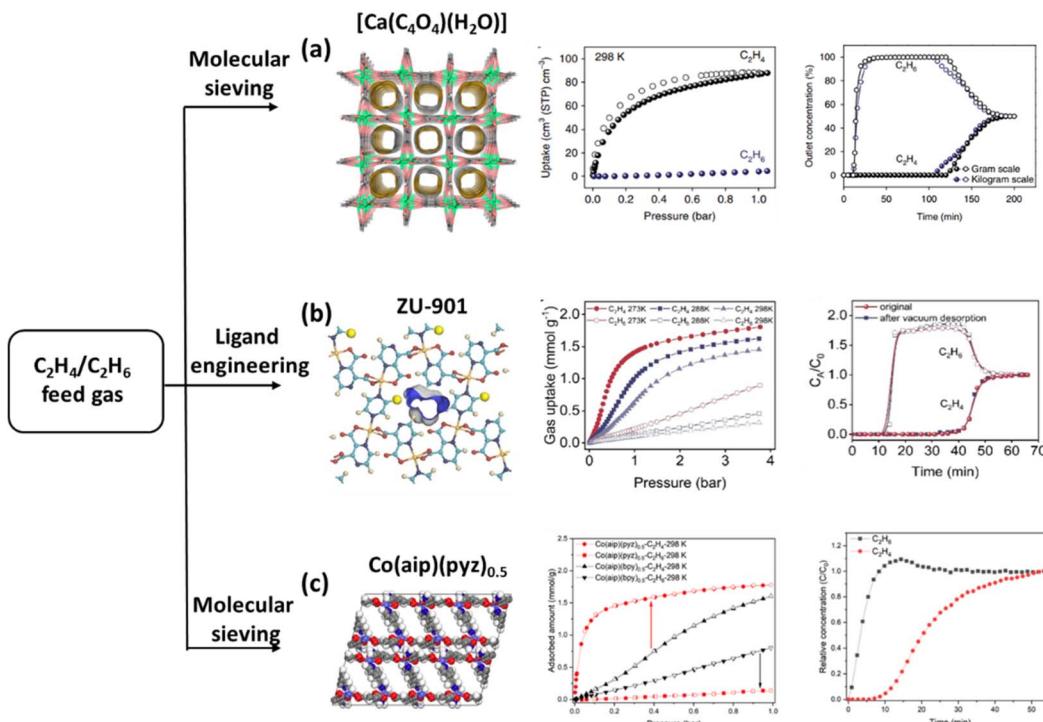


Fig. 12 (a) The crystal structure of guest-free UTSA-280, single-component sorption isotherms of C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> at 298 K and breakthrough curves from different scales for an equimolar C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> mixture at 298 K and 1 bar. (b) The internal structures of ZU-901, adsorption isotherms of C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> at different temperatures and high pressure, and breakthrough curve of the fresh and regenerated sample via vacuum desorption. (c) The structures of Co(aip)(pyz)<sub>0.5</sub>, C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> adsorption isotherms of Co(aip)(pyz)<sub>0.5</sub> at 298 K, and experimental column breakthrough curves for C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> equimolar binary mixtures at 298 K. Reprinted with permission from ref. 70. Copyright 2018 Springer Nature Limited, ref. 116. Copyright 2023 John Wiley & Sons, ref. 117. Copyright 2023 Elsevier Inc, respectively.

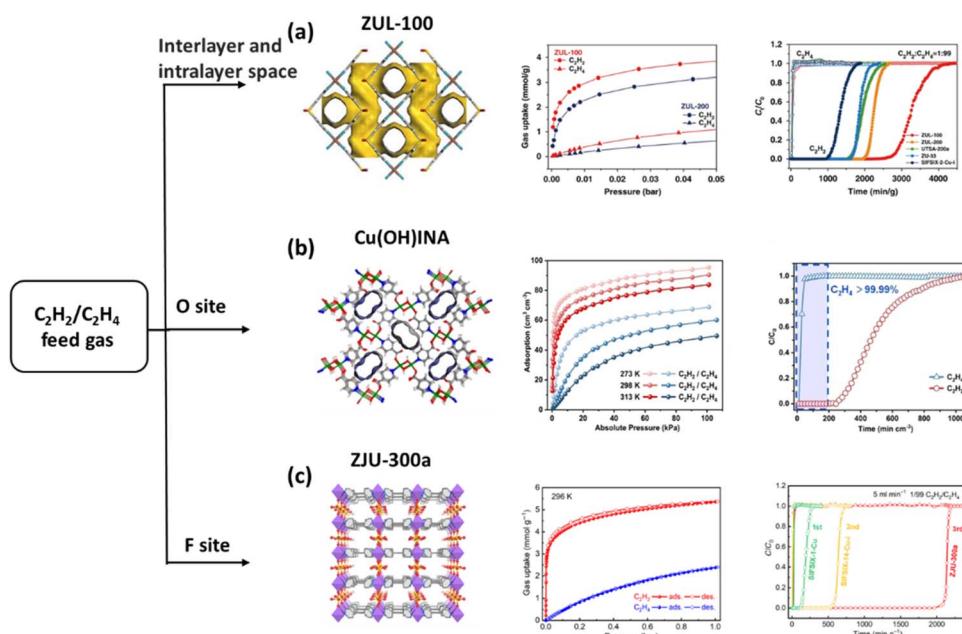


Fig. 13 (a) Pore geometry of ZUL-100 with an intralayer, C<sub>2</sub>H<sub>2</sub> and C<sub>2</sub>H<sub>4</sub> adsorption isotherms at 298 K, and experimental column breakthrough curves for the C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> mixture at 298 K and 1 bar. (b) Structure and one-dimensional channel of Cu(OH)INA, adsorption isotherms of C<sub>2</sub>H<sub>2</sub> and C<sub>2</sub>H<sub>4</sub> at various temperatures, and breakthrough experiment with a C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> mixture at a flow rate of 2 mL min<sup>-1</sup> at 298 K. (c) Single-crystal structure of ZJU-300, adsorption isotherms for C<sub>2</sub>H<sub>2</sub> and C<sub>2</sub>H<sub>4</sub> at 298 K, and experimental column breakthrough curves for 1/99 C<sub>2</sub>H<sub>2</sub>/C<sub>2</sub>H<sub>4</sub> separation at a flow rate of 5 mL min<sup>-1</sup> under ambient conditions. Reprinted with permission from ref. 119. Copyright 2020 Springer Nature Limited, ref. 120. Copyright 2024 John Wiley & Sons, ref. 121. Copyright 2023 The American Association for the Advancement of Science, respectively.



$\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  separation selectivity (7.1 to 10.6). These values reflected the emerging separation performance of  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$ . In the breakthrough experiments, highly-pure  $\text{C}_2\text{H}_4$  broke the sample beds of SIFSIX1-Cu and SIFSIX2-Cu-i at first, while 0.38 and 0.73 mmol g<sup>-1</sup> of  $\text{C}_2\text{H}_2$  was captured by from the 1/99 mixture, respectively. After replacing the linear ligand from a 2-c to 4-c linker, ZJU-300 with densely decorated  $\text{SiF}_6^{2-}$  ions also exhibited a promising  $\text{C}_2\text{H}_2$  uptake of 3.23 mmol g<sup>-1</sup> (0.01 bar, 296 K). This result is substantially higher than that of all the SIFSIX materials (Fig. 13c).<sup>121</sup>

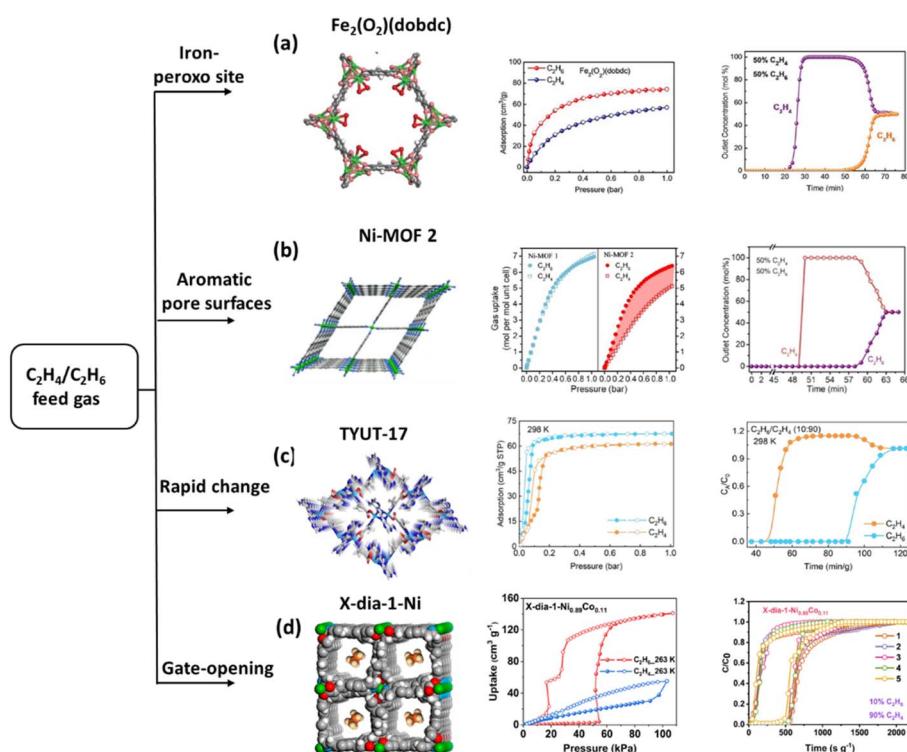
When the pressure was increased to 1 bar, the  $\text{C}_2\text{H}_2$  uptake increased to 5.4 mmol g<sup>-1</sup>, while the  $\text{C}_2\text{H}_4$  uptake was only 2.39 mmol g<sup>-1</sup>. It is noteworthy that the  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  (1/99) selectivity is as high as 1672. This observation was subsequently confirmed through dynamic breakthrough experiments, which revealed that the  $\text{C}_2\text{H}_4$  productivity reached 436.7 mmol g<sup>-1</sup>. Moreover, the material maintained good stability under humidity and an acidic gas atmosphere.

**3.2.2  $\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  feed gas.**  $\text{Fe}_2(\text{O}_2)(\text{dobdc})$  demonstrates a superior  $\text{C}_2\text{H}_6$  (74.3 cm<sup>3</sup> g<sup>-1</sup>) adsorption capacity in comparison to that of  $\text{C}_2\text{H}_4$  (57.3 cm<sup>3</sup> g<sup>-1</sup>) at 298 K and 1 bar, providing an increased  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (v/v: 1/1) selectivity (from 3.5 to 4.4) under increased pressure.<sup>46</sup> Thus, in the breakthrough experiment, the  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixture can be completely separated to

yield 99.95% pure  $\text{C}_2\text{H}_4$ , up to 2172 mmol per liter ( $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$ , 50/50) and 6855 mmol per liter ( $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$ , 10/90). Importantly, the material can be readily regenerated for cycling separations. Encouraged by this finding, breakthrough experiments were performed on multi-components. The highly efficient separation performance suggests that  $\text{Fe}_2(\text{O}_2)(\text{dobdc})$  is an effective separator for  $\text{C}_2\text{H}_4$  purification when the concentration of  $\text{C}_2\text{H}_6$  is low, as well as the existence of  $\text{CH}_4$ ,  $\text{H}_2$ , and  $\text{C}_2\text{H}_2$  impurities (Fig. 14a).

The two isostructural MOFs Ni-MOF-1 and Ni-MOF-2 have also been investigated for  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  separation.<sup>63</sup> For Ni-MOF-1, nearly the same adsorption isotherms of  $\text{C}_2\text{H}_6$  (113 cm<sup>3</sup> g<sup>-1</sup>) and  $\text{C}_2\text{H}_4$  (116 cm<sup>3</sup> g<sup>-1</sup>) were observed at 298 K and 1 bar. However, Ni-MOF 2 demonstrated a substantially higher adsorption of  $\text{C}_2\text{H}_6$  (133 cm<sup>3</sup> g<sup>-1</sup>) over  $\text{C}_2\text{H}_3$  (105 cm<sup>3</sup> g<sup>-1</sup>). Along with an uptake difference of 40 cm<sup>3</sup> g<sup>-1</sup> at 0.5 bar, Ni-MOF 2 can efficiently separate the equimolar  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixture gas in dynamic penetration experiments, yielding a  $\text{C}_2\text{H}_4$  productivity of 12 L kg<sup>-1</sup>. In addition, cycling experiments demonstrated that Ni-MOF-2 has good reproducibility for this separation (Fig. 14b).

Different from the previously mentioned rigid MOFs, TYUT-17, a flexible-robust framework, exhibited a selective adsorption tendency for  $\text{C}_2\text{H}_6$ .<sup>66</sup> TYUT-17 demonstrated a higher



**Fig. 14** (a) Structures of  $\text{Fe}_2(\text{O}_2)(\text{dobdc})$ , adsorption isotherms of  $\text{C}_2\text{H}_6$  and  $\text{C}_2\text{H}_4$  at 298 K, and experimental column breakthrough curves for a  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixture at 298 K and 1.01 bar. (b) The three-dimensional open framework of Ni-MOF 2,  $\text{C}_2\text{H}_4$  and  $\text{C}_2\text{H}_6$  single-component adsorption isotherms at 298 K, and dynamic breakthrough curves of equimolar  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  gas mixtures at 298 K and 1 bar. (c) The structure of TYUT-17, single-component gas adsorption isotherms at 298 K, and breakthrough curves of TYUT-17 for  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (v/v, 10/90) mixtures at 298 K and 1 bar. (d)  $\text{C}_2\text{H}_6$  adsorption induced the structural transformation of X-dia-1-Ni,  $\text{C}_2\text{H}_4$  and  $\text{C}_2\text{H}_6$  adsorption isotherms for X-dia-1-Ni<sub>0.89</sub>Co<sub>0.11</sub> at 273 K, and cyclic breakthrough separation experiments for  $\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  (1/9) mixtures performed at 100 kPa and 263 K. Reprinted with permission from ref. 46. Copyright 2018 The American Association for the Advancement of Science, ref. 63. Copyright 2023 John Wiley & Sons, ref. 66. Copyright 2024 John Wiley & Sons, ref. 56. Copyright 2024 American Chemical Society, respectively.



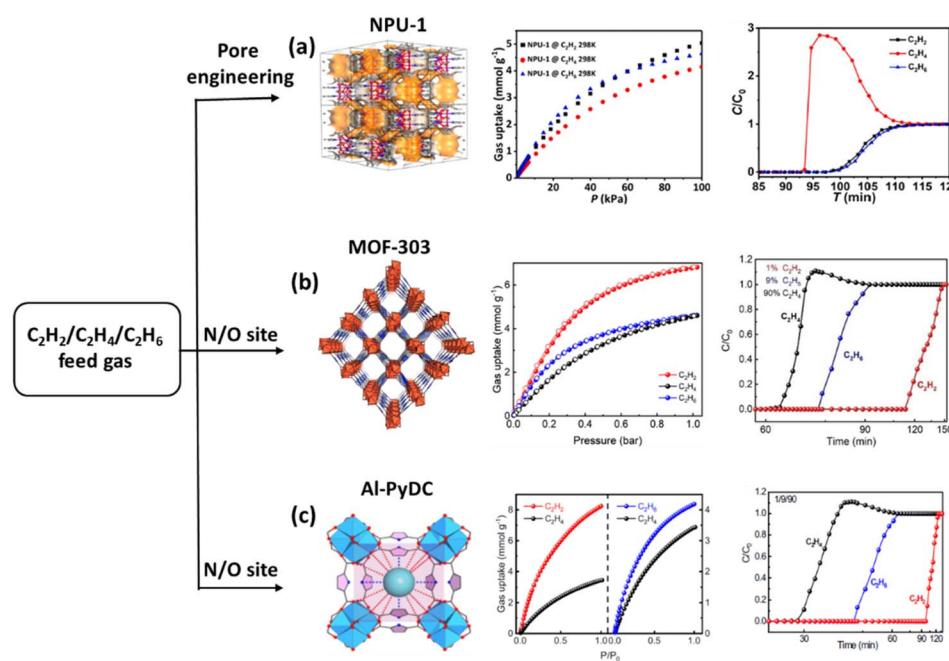
adsorption capacity of  $62.6 \text{ cm}^3 \text{ g}^{-1}$  for  $\text{C}_2\text{H}_6$ , at 298 K and 0.1 bar; however, the  $\text{C}_2\text{H}_4$  uptake was found to be  $18.7 \text{ cm}^3 \text{ g}^{-1}$ , providing an exceptional  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  selectivity (6.4) and uptake ratio (3.3). It was observed that an increase in temperature resulted in a shift of the gate-opening phenomenon to a high-pressure region, but the  $P_{\text{C}_2\text{H}_4}$  always lagged behind the  $P_{\text{C}_2\text{H}_6}$ , indicating the reality of sample regeneration. Following the introduction of a feed gas comprising  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures to the packed TYUT-17, the observation of different retention times indicated unambiguous separation. The productivity of  $\text{C}_2\text{H}_4$  of >99.9% purity was recorded at  $77.4 \text{ L kg}^{-1}$ . Furthermore, the gate-opening of TYUT-17 facilitated the separation of the  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixture at 313 K (Fig. 14c).

To promote the separation temperature, an implication of saving energy in the separation process, we developed two hydrogen-bonded interpenetrated frameworks, NTU-101 and NTU-101-NH<sub>2</sub>.<sup>62</sup> A three H-bond linked interpenetrated porous framework, NTU-101 exhibited a structural response to  $\text{C}_2\text{H}_6$  and  $\text{C}_2\text{H}_4$  in the temperature range of 288 to 303 K. Comparably, the single H-bond connected framework of NTU-101-NH<sub>2</sub> exhibited a structural response to  $\text{C}_2\text{H}_6$  and  $\text{C}_2\text{H}_4$  in a wider temperature range of 288 to 333 K, particularly the gate-opening pressure under  $\text{C}_2\text{H}_6$  is always earlier than that of  $\text{C}_2\text{H}_4$ . Consequently, NTU-101-NH<sub>2</sub> is capable of separating the  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (1/9, v/v) mixtures at a relative higher temperature of 328 K. To our knowledge, this is the first example that can selectively capture  $\text{C}_2\text{H}_6$  from  $\text{C}_2\text{H}_4$  at such a higher temperature and yielded a productivity of polymer-grade  $\text{C}_2\text{H}_4$ , of up to 15.7 mL

$\text{g}^{-1}$ . Meanwhile, the robust NTU-101-NH<sub>2</sub> can be prepared on a large scale through the stirring method at room temperature.

In addition, the flexible dia coordination network of X-dia-1-Ni<sub>0.89</sub>Co<sub>0.11</sub> exhibited preferred gate-opening for  $\text{C}_2\text{H}_6$ , but not for  $\text{C}_2\text{H}_4$  at 263 K.<sup>56</sup> The unique structural change driven the  $\text{C}_2\text{H}_6$  adsorption sharply increased at a pressure of 50 kPa, and the maximum adsorption capacity reached  $131.4 \text{ cm}^3 \text{ g}^{-1}$ . However, the structure did not show evident gate-opening, thus providing a low  $\text{C}_2\text{H}_4$  uptake of  $44.8 \text{ cm}^3 \text{ g}^{-1}$ . The ability for harvesting high-purity (99.9%)  $\text{C}_2\text{H}_4$  directly was validated by dynamic breakthrough experiments at 263 K ( $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$ : 1/9, v/v) (Fig. 14d).

**3.2.3  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  feed gas.** The harvesting of highly pure  $\text{C}_2\text{H}_4$  from the ternary mixture of  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  requires the MOFs to have the ability to selectively adsorb  $\text{C}_2\text{H}_2$  and  $\text{C}_2\text{H}_6$  simultaneously. An earlier example of this was exhibited by TJJ-100, due to the hierarchy of weak sorbent–sorbate interactions.<sup>82</sup> To promote the separation performance, three isostructural MOFs, named NPU-1, NPU-2 and NPU-3, were synthesized from an Mn<sub>6</sub> cluster, rigid dicarboxylate ligands and a 3-connected pyridyl-based tritopic ligand.<sup>122</sup> With the smallest pore size among the three, NPU-1 exhibited the largest uptakes for  $\text{C}_2\text{H}_2$  ( $5.1 \text{ mmol g}^{-1}$ ),  $\text{C}_2\text{H}_6$  ( $4.5 \text{ mmol g}^{-1}$ ) and  $\text{C}_2\text{H}_4$  ( $4.2 \text{ mmol g}^{-1}$ ). The adsorption selectivities for  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (1/1, v/v) and  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  (1/1, v/v) were calculated to be 1.32 and 1.4, respectively. The preferential uptake of  $\text{C}_2\text{H}_2$  and  $\text{C}_2\text{H}_6$  allows this MOF to produce highly pure  $\text{C}_2\text{H}_4$  from equimolar ternary mixtures in one-step. The stability of this separation was



**Fig. 15** (a) NPU-1 three-dimensional structure; C<sub>2</sub> gas sorption isotherms of NPU-1 at 298 K; and experimental breakthrough curves at 298 K for C<sub>2</sub> (1 : 1 : 1 mixture) separation. (b) The framework of MOF-303, C<sub>2</sub> adsorption isotherm of MOF-303 at 296–313 K, and experimental column breakthrough curves of MOF-303 for C<sub>2</sub> mixtures under ambient conditions. (c) Multiple supramolecular binding sites in the pores of Al-PyDC, C<sub>2</sub> adsorption isotherms of Al-PyDC at 296 K, and experimental column breakthrough curves of Al-PyDC for C<sub>2</sub> mixtures under ambient conditions. Reprinted with permission from ref. 122. Copyright 2021 American Chemical Society, ref. 123. copyright 2023 John Wiley & Sons, ref. 83. Copyright 2023 Springer Nature Limited, respectively.



validated through cycling experiments. Based on the packing of the  $\text{C}_2\text{H}_2$ -selective MOF and  $\text{C}_2\text{H}_6$ -selective MOF, the same group developed a synergistic sorbent separation technology for purification of  $\text{C}_2\text{H}_4$  from  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  mixtures.<sup>88</sup> This technology may pave the way for the development of new separation systems based on reported MOFs for challenging separations (Fig. 15a).

The aluminium-based MOF (MOF-303) is a stable, low-cost, and easily scalable separation material.<sup>123</sup> It also demonstrated preferred  $\text{C}_2\text{H}_2$  ( $7.94 \text{ mmol g}^{-1}$ ) and  $\text{C}_2\text{H}_6$  ( $5.01 \text{ mmol g}^{-1}$ ) adsorption compared with that of  $\text{C}_2\text{H}_4$ . In the dynamic breakthrough experiment, the binary mixture  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  (1/99, v/v) can be separated, with an outlet effluent  $\text{C}_2\text{H}_4$  yield of  $5.07 \text{ mmol g}^{-1}$  and a purity greater than 99.95%. Furthermore, the yield of the  $\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  binary mixture is also as high as  $0.56 \text{ mmol g}^{-1}$ , and the purity is greater than 99.95%. In the ternary mixture (1/9/90, v/v/v), the productivity of  $\text{C}_2\text{H}_4$  is as high as  $1.35 \text{ mmol g}^{-1}$  (>99.95%), exhibiting excellent separation performance (Fig. 15b).

The low polarity O/N supramolecular binding sites allow Al-PyDC to show both higher  $\text{C}_2\text{H}_2$  ( $8.24 \text{ mmol g}^{-1}$ ) and  $\text{C}_2\text{H}_6$  ( $4.20 \text{ mmol g}^{-1}$ ) uptakes in comparison to  $\text{C}_2\text{H}_4$  ( $3.44 \text{ mmol g}^{-1}$ ) at 296 K and 1 bar,<sup>83</sup> along with both a good  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  (1/99, v/v) selectivity of 4.3 and  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (50/50, v/v) selectivity of 1.9. In the dynamic breakthrough experiments, both  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$  and  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  binary mixtures can be separated. Therefore, Al-PyDC can separate the  $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  ternary mixtures for getting polymer-grade  $\text{C}_2\text{H}_4$  in one-step with a productivity of  $1.61 \text{ mmol g}^{-1}$ . Furthermore, the system demonstrates consistent recyclability over 20 cycles, exhibiting no decline in performance under acidic gas conditions (Fig. 15c).

### 3.3 $\text{C}_3\text{H}_6$ harvesting from the desorption process

**3.3.1  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  feed gas.** The harvesting of pure  $\text{C}_3\text{H}_6$  at the desorption process from  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  feed gas requires a MOF that exhibits superior  $\text{C}_3\text{H}_6$  uptake attributable to either stronger interaction sites or a favoured  $\text{C}_3\text{H}_6$  trap.<sup>109,124,125</sup> Typically, based on the sieving effect, KAUST-7 exhibits specific adsorption of  $\text{C}_3\text{H}_6$  but did not permit  $\text{C}_3\text{H}_8$  to diffuse/adsorb into the pore system at 298 K up to 1 bar, as evidenced by fully overlaid adsorption isotherms between pure  $\text{C}_3\text{H}_6$  and equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$ .<sup>32</sup> Breakthrough experiments further demonstrated that KAUST-7 flowed directly out of the packed

column, while  $\text{C}_3\text{H}_6$  penetrated after  $\sim 480$  s, followed by further desorption to obtain pure  $\text{C}_3\text{H}_6$ , and the recovery is  $\sim 2 \text{ mol per kg per hour}$  at standard ambient temperature and pressure. In addition, KAUST-7 possesses a noticeably high chemical (tolerance to water vapor and hydrogen sulfide) and thermal stability (Fig. 17a). Another example of sieving separation is ZU-609 with local sieving channels, which has the potential to separate  $\text{C}_3\text{H}_6$  from  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures due to the presence of molecular sieving gates and fast diffusion channels in the local sieving channels and the similar size of the molecular sieving gates to the size of the  $\text{C}_3\text{H}_6$  molecules.<sup>33</sup> The single-component adsorption isotherms indicated sieving performance for  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  with an uptake ratio of 22.3 at 298 K and 1 bar. Breakthrough experiments confirmed the superior adsorption performance of ZU-609 on  $\text{C}_3\text{H}_6$  with a dynamic adsorption capacity of about  $1.64 \text{ mmol g}^{-1}$  and a yield of  $32.2 \text{ L kg}^{-1}$  for obtaining high-purity (99.9%)  $\text{C}_3\text{H}_6$  (Fig. 17f).

In the presence of an orthogonal-array dynamic molecular sieve, both stepwise  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  adsorption isotherms were observed in JNU-3a.<sup>65</sup> The rapid uptake at low pressure gradually diminished and the step shifted to higher pressures following increased pressure, a phenomenon attributable to the local dynamic of the molecular pocket. Importantly, at an earlier gate opening pressure, the  $\text{C}_3\text{H}_6$  uptake reached  $58.6 \text{ cm}^3 \text{ g}^{-1}$  at 303 K and 1 bar, which is considerably than that of  $\text{C}_3\text{H}_8$  ( $42.5 \text{ cm}^3 \text{ g}^{-1}$ ). The density of  $\text{C}_3\text{H}_6$  was calculated to be  $404 \text{ g L}^{-1}$  based on the pore volume, reflecting its good packing ability of  $\text{C}_3\text{H}_6$  compared to the density of gaseous  $\text{C}_3\text{H}_6$  ( $1.707 \text{ g L}^{-1}$ ). Breakthrough curves revealed that JNU-3a can separate equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  at 303 K.  $\text{C}_3\text{H}_8$  was eluted from the packed column at about 70 min, while  $\text{C}_3\text{H}_6$  penetrated through the column after 35 min. In the desorption stage, JNU-3a was able to obtain high-purity  $\text{C}_3\text{H}_6$  with a productivity of  $34.2 \text{ L kg}^{-1}$  (Fig. 17b).

In the context of quasi-discrete cavities,  $\text{DL-mal-MOF}$  demonstrated rapid and different uptakes of  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  in the low pressure region, with uptake values of up to  $1.68 \text{ mmol g}^{-1}$  and  $1.21 \text{ mmol g}^{-1}$  at 298 K and 1 kPa.<sup>27</sup> In contrast, both  $\text{L-mal-MOF}$  and  $\text{D-mal-MOF}$  exhibited nearly the same adsorption isotherms for  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$ . Remarkably,  $\text{DL-mal-MOF}$  demonstrated both a high Henry's selectivity of 5.86 and kinetic selectivity of 114.2. The observed disparity in diffusion rates facilitates the dynamic separation of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures (50/50, or 95/5, v/v) at 298 K. During the desorption stage, the polymer-grade  $\text{C}_3\text{H}_6$  was attained, with a yield of  $2.1 \text{ L kg}^{-1}$  (Fig. 17c).

Dependent on the cooperative global and local dynamics, FDC-4 exhibited gate-opening during  $\text{C}_3\text{H}_6$  adsorption, yielding an uptake of  $122 \text{ cm}^3 \text{ g}^{-1}$  at 300 K.<sup>74</sup> However,  $\text{C}_3\text{H}_8$  adsorption in this narrow channel adopts a diffusion controlled manner, thereby resulting in the very low uptake of  $\text{C}_3\text{H}_8$  being observed. The substantial differences in both uptake capacity and adsorption kinetics enabled the dynamic separation of an equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixture, thereby revealing sieving performance and producing  $\text{C}_3\text{H}_6$  ( $19.5 \text{ L kg}^{-1}$ ) with a purity of 99.7% (Fig. 17d).

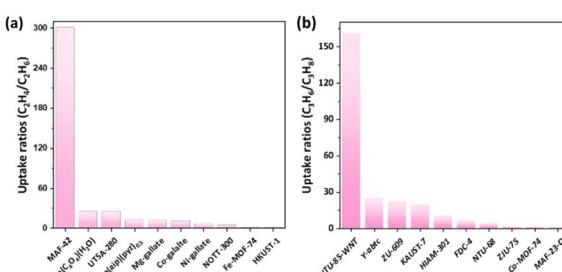


Fig. 16 The top ten in terms of uptake ratios for  $\text{C}_2\text{H}_4/\text{C}_2\text{H}_6$  (a) and  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  (b).



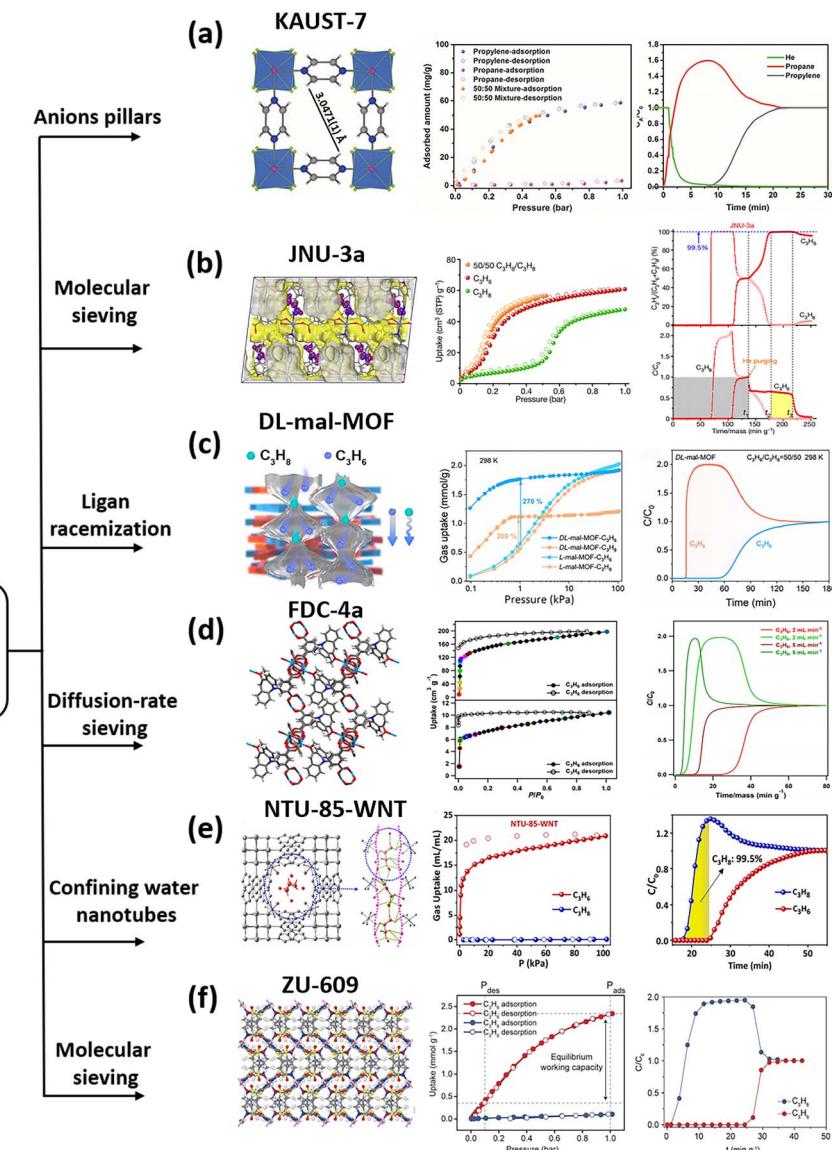


Fig. 17 (a) The crystal structure of KAUST-7, adsorption isotherms for pure  $\text{C}_3\text{H}_8$ , pure  $\text{C}_3\text{H}_6$ , and  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  equimolar mixtures at 298 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 298 K. (b) The crystal structure of  $\text{C}_3\text{H}_6$ @JNU-3a, adsorption isotherms for pure  $\text{C}_3\text{H}_8$ , pure  $\text{C}_3\text{H}_6$ , and  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  equimolar mixtures at 303 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 303 K. (c) The ligand racemization strategy is illustrated, adsorption isotherms of  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  on DL-mal-MOF and breakthrough experimental curves of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  (50/50, v/v) mixtures at 298 K. (d) The crystal structure of FDC-4a, adsorption isotherms for  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  at 240 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 300 K. (e) The crystal structure of NTU-85-WNT $\supset$  $\text{C}_3\text{H}_6$ , hydrogen bond interactions of  $\text{C}_3\text{H}_6$  in one-dimensional nanochannels, adsorption isotherms for  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  at 298 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 300 K. (f) The crystal structure of ZU-609, adsorption isotherms for pure  $\text{C}_3\text{H}_8$  and  $\text{C}_3\text{H}_6$  at 298 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 298 K and 1 bar. Reprinted with permission from ref. 32. Copyright 2016 Science, ref. 65. Copyright 2021 Springer Nature Limited, ref. 27. Copyright 2023 American Chemical Society, ref. 74. Copyright 2024 Springer Nature Limited, ref. 37. Copyright 2023 American Chemical Society, ref. 33. Copyright 2024 Science, respectively.

NTU-85-WNT, characterized by localized dynamics of a water nanotube within a rigid framework, exhibited rapid  $\text{C}_3\text{H}_6$  uptake within the initial pressure range, and reached 20.9  $\text{mL mL}^{-1}$ .<sup>37</sup> Conversely,  $\text{C}_3\text{H}_8$  uptake was observed to be negligible (0.13  $\text{mL mL}^{-1}$ ). The clear cut-off adsorption yields an extremely high adsorption selectivity of 1570 at 298 K and 1 bar. This sieving phenomenon is very similar to that of the rigid separator of KAUST-1, but the underlying mechanism is totally different. Breakthrough experiments substantiated the exclusive

adsorption of  $\text{C}_3\text{H}_6$  by NTU-85-WNT, with  $\text{C}_3\text{H}_8$  eluting out of the packed column initially. The confined effect of the water nanotube enables the harvesting of adsorbed  $\text{C}_3\text{H}_6$  at the desorption stage with a high purity of 98.8%. However, the lower porosity of the water nanotube limits the  $\text{C}_3\text{H}_6$  productivity (1.6  $\text{mL mL}^{-1}$ ) at a certain content (Fig. 17e).

The same group developed a new concept of delicate softness in a temperature-responsive MOF (NTU-68) for efficient separation of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  in the next year.<sup>40</sup> Quantitatively, NTU-68



exhibited a pronounced uptake of  $C_3H_6$  ( $33.3 \text{ cm}^3 \text{ g}^{-1}$  at 298 K) at 2 kPa, which exceeds the total uptake of KAUST-7 at 1 bar. The calculated density of  $C_3H_6$  is as high as  $565.9 \text{ mL mL}^{-1}$ , closely approaching the density of liquid  $C_3H_6$  ( $606 \text{ mL mL}^{-1}$ ). In contrast, the MOF exhibited a markedly lower  $C_3H_8$  uptake ( $7.5 \text{ cm}^3 \text{ g}^{-1}$ ) at 298 K and 1 bar. As illustrated by the changing trend (in the temperature range of 273 to 323 K), the temperature-responsive delicate softness around the channel neck results in a thermodynamic preferred  $C_3H_6$  adsorption (the higher the temperature, the lower the uptake), but a kinetic dominated  $C_3H_8$  uptake (the higher the temperature, the higher the uptake). Consequently, the breakthrough separation efficiency of this mixture was found to be twofold enhanced upon increasing the temperature from 273 to 298 K. Notably, the separation process remained unaffected by the presence of general impurities.

The ftw-type MOFs consisting of a hexanuclear cluster M6 ( $M = Zr^{4+}, Hf^{4+}, Ce^{4+}, Y^{3+}$ ) and a four-coordination linker have potential for molecular sieving due to the large cage-like cavities interconnected by narrow windows. For example, the iso- configurational MOFs ftw-MOF-ABTC and HIAM-301 both exhibit excellent molecular sieving in the separation of  $C_3H_6/C_3H_8$  mixtures, but the mechanisms of separation are different.<sup>64,126</sup> The MOF-ABTC achieves molecular sieving by adjusting the length of the ligand only to change the window size and thus its pore size. In contrast, HIAM-301 has a highly distorted pore due to the inconsistency of the plane created by the mutual rotation of the octahedra and the large aspect ratio of the  $H_4$ eddi linker; in addition, the use of  $Y_6$  clusters instead of  $Zr_6$  clusters changes the framework from electrically neutral to an anionic framework and the presence of counterions in the cavities provides additional pore size modulation. The two synergistically regulate the pore size of HIAM-301 to achieve efficient sieving of  $C_3H_6/C_3H_8$ . In terms of separation performance, HIAM-301 showed better separation performance

compared to ftw-MOF-ABTC, with  $3.16 \text{ mmol g}^{-1}$  adsorption for  $C_3H_6$  and negligible adsorption for  $C_3H_8$  ( $<0.3 \text{ mmol g}^{-1}$ ) at 298 K and 1 bar, enabling the complete separation of  $C_3H_6$  and  $C_3H_8$ . Breakthrough experiments at room temperature also showed that the dynamic adsorption capacity of HIAM-301 for  $C_3H_6$  is  $46.4 \text{ cm}^3 \text{ g}^{-1}$ , which enabled the production of polymer-grade  $C_3H_6$ . The single-component adsorption curves and breakthrough experiments of ftw-MOF-ABTC for  $C_3H_6$  and  $C_3H_8$  are not significant ( $2.3 \text{ mmol g}^{-1}$ ,  $2.46 \text{ mmol g}^{-1}$ ), but the adsorption kinetics of  $C_3H_6$  is faster than that of  $C_3H_8$  under the low-pressure condition, so that  $C_3H_6/C_3H_8$  is effectively separated in the breakthrough experiment (Fig. 18a and b). Furthermore, we also compared the top ten  $C_3H_6/C_3H_8$  adsorption ratios, as shown in Fig. 16b, where NTU-85-WNT reaches an adsorption ratio of 160, which has the potential to completely separate the  $C_3H_6/C_3H_8$  mixture.<sup>37</sup> This summary provides a reference for efficient light hydrocarbon separation.

In addition, other MOFs, such as ZJU-75, MAF-23-O, Y-abtc, and Co-MOF-74, have been observed to exhibit preferential adsorption of  $C_3H_8$  from mixtures containing  $C_3H_6$ .<sup>6,22,88,90</sup> However, due to the closed separation mechanism discussed above, we will not discuss them further.

**3.3.2  $C_3H_4(CH_3C\equiv CH)/C_3H_6/C_3H_8$  feed gas.** The purification of  $C_3H_6$  from ternary or quaternary C3 hydrocarbons has been considered as the most challenging process in gas separation.<sup>127</sup> To recognize the extremely small molecular difference, we developed a group of soft PCPs (NTU-65-FeZr, NTU-65-FeTi, NTU-65-CoZr and NTU-65-CoTi)<sup>51</sup> on the basis of a global soft framework of NTU-65 that has a  $Cu^{2+}$  node and electro-negative site of a  $SiF_6^{2-}$  anion. The changed strength of the coordination bonds, derived from the changed metal node and the multiple sites, endowed the framework with the capacity to show a system structure response: from solely the  $C_3H_4$  response at 17.1 kPa in NTU-65-FeZr to the sequential response

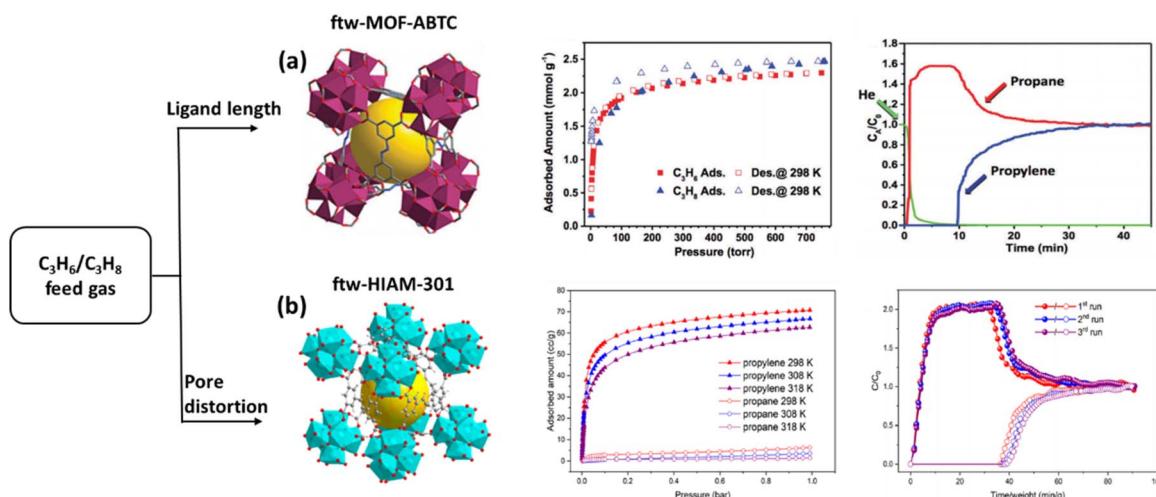


Fig. 18 (a) The crystal structure of ftw-MOF-ABTC, adsorption isotherms for pure  $C_3H_8$  and  $C_3H_6$  at 298 K, and breakthrough experimental curves of equimolar  $C_3H_6/C_3H_8$  mixtures at 298 K. (b) The crystal structure of ftw-HIAM-301, adsorption isotherms for pure  $C_3H_8$  and  $C_3H_6$  at different temperatures, and breakthrough experimental curves of equimolar  $C_3H_6/C_3H_8$  mixtures at room temperature. Reprinted with permission from ref. 126. Copyright 2018. The Royal Society of Chemistry, ref. 64. Copyright 2021 American Chemical Society, respectively.



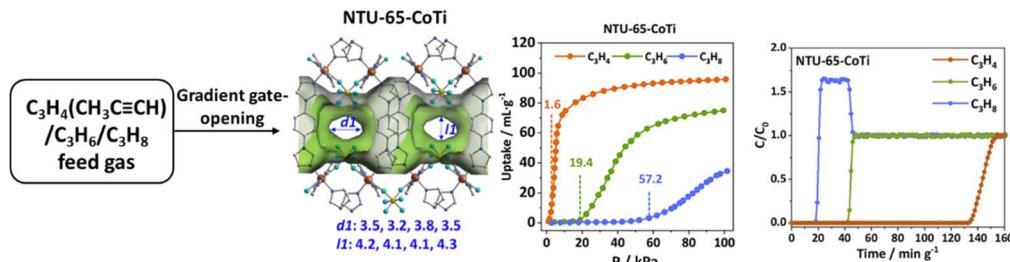


Fig. 19 The crystal structure of NTU-65-CoTi, adsorption isotherms for  $\text{C}_3\text{H}_4$ ,  $\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_8$  at 273 K, and breakthrough experimental curves of  $\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  (0.5/49.75/49.75, v/v/v, 2  $\text{mL min}^{-1}$ ) mixtures at 273 K. Reprinted with permission from ref. 51. Copyright 2024 The Royal Society of Chemistry.

of  $\text{C}_3\text{H}_4$  at 1.6 kPa,  $\text{C}_3\text{H}_6$  response at 19.4 kPa and finally  $\text{C}_3\text{H}_8$  at 57.2 kPa in NTU-65-CoTi at 273 K. Therefore, such unprecedented gradient structural-response driven NTU-65-CoTi is capable of separating  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  (0.5/49.75/49.75, v/v/v, 2  $\text{mL min}^{-1}$ ) mixtures. Based on breakthrough results, two beds were connected for the adsorption cycle. From bed B,  $\text{C}_3\text{H}_6$  with a purity of 99.5% can be harvested at the desorption stage, while  $\text{C}_3\text{H}_8$  with highly purity can be obtained at the adsorption stage (Fig. 19).

### 3.4 $\text{C}_3\text{H}_6$ harvesting from the adsorption process

#### 3.4.1 $\text{C}_3\text{H}_4$ ( $\text{CH}_3\text{C}\equiv\text{CH}$ and/or $\text{CH}_2=\text{C}=\text{CH}_2$ )/ $\text{C}_3\text{H}_6$ feed gas.

To remove the trace amount of  $\text{C}_3\text{H}_4$  from  $\text{C}_3\text{H}_6$ , a stronger

host– $\text{C}_3\text{H}_4$  interaction is imperative. This is necessary to ensure that  $\text{C}_3\text{H}_6$ , in its highly pure state, can be effectively harvested from the adsorption process. Typically, ZNU-2, an ultra-stable MOF with cage-like pores (decorated with  $\text{TiF}_6^{2-}$  anions) interconnected with each other by contracted necks, has been reported for selective  $\text{C}_3\text{H}_4$  capture from  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6$  mixtures. Single gas adsorption isotherms revealed that ZNU-2 demonstrated both high  $\text{C}_3\text{H}_4$  uptake at 0.01 bar (3.9  $\text{mmol g}^{-1}$ ) and 1 bar (7.7  $\text{mmol g}^{-1}$ ) at 298 K, while the uptake of  $\text{C}_3\text{H}_6$  reached 0.61 and 5.3  $\text{mmol g}^{-1}$ , respectively. Therefore, with the combination of a  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6$  selectivity (1/99, v/v) of 12.5, the separation potential, a concept developed by Krishna, of ZNU-2 is as high as 31.0  $\text{mmol g}^{-1}$ , the maximum value among the

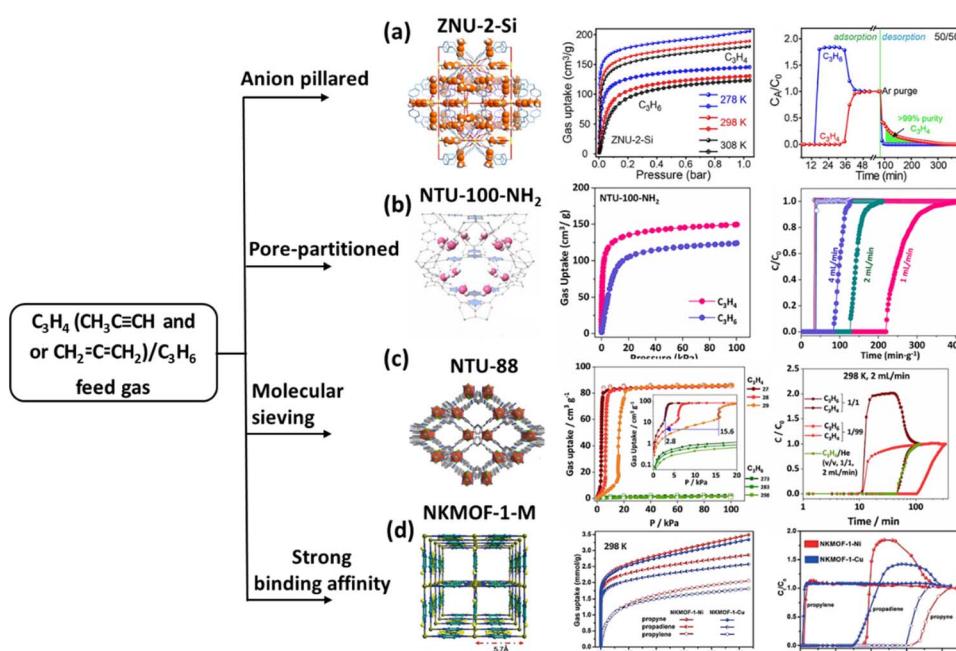


Fig. 20 (a) The crystal structure of  $\text{C}_3\text{H}_4$  loaded ZNU-2-Si,  $\text{C}_3\text{H}_4$  and  $\text{C}_3\text{H}_6$  adsorption isotherms for ZNU-2-Si at 278, 298 and 308 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6/\text{C}_3\text{H}_8$  mixtures at 298 K. (b) The crystal structure of NTU-100-NH<sub>2</sub>,  $\text{C}_3\text{H}_4$  and  $\text{C}_3\text{H}_6$  adsorption isotherms for ZNU-2-Si at 278, 298 and 308 K, and breakthrough experimental curves of equimolar  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6$  mixtures at different inlet rates at 298 K. (c) The crystal structure of NTU-88, adsorption isotherms for  $\text{C}_3\text{H}_4$  and  $\text{C}_3\text{H}_6$  at different temperatures, and breakthrough curves of NTU-88 for  $\text{C}_3\text{H}_4/\text{C}_3\text{H}_6$  and  $\text{C}_3\text{H}_4/\text{He}$  at 298 K. (d) The 3D structure of NKMOF-1-M (M = Ni or Cu), adsorption isotherms for  $\text{C}_3\text{H}_4$ ,  $\text{CH}_2=\text{C}=\text{CH}_2$  and  $\text{C}_3\text{H}_6$  and breakthrough experimental curves of  $\text{C}_3\text{H}_4$ ,  $\text{CH}_2=\text{C}=\text{CH}_2$  and  $\text{C}_3\text{H}_6$  (0.5/0.5/99, v/v/v) mixtures at 298 K. Reprinted with permission from ref. 128. Copyright 2023 The Royal Society of Chemistry, ref. 84. Copyright 2024 John Wiley & Sons, ref. 69. Copyright 2023 John Wiley & Sons, ref. 52. Copyright 2019 John Wiley & Sons, respectively.



MOFs. After the introduction of the  $C_3H_4/C_3H_6$  (1/99, v/v) feed gas into the packed column, pure  $C_3H_6$  broke out at first, while  $C_3H_4$  was detected about 150 min later. The clear separation interval indicates a productivity of  $C_3H_6$  of 42.0 mol kg<sup>-1</sup>. Similarly, an anionic pillared caged MOF, ZNU-2-M (M = Si, Ti, Nb) was also developed for light hydrocarbon separations (Fig. 20a).<sup>128</sup>

NTU-100-NH<sub>2</sub>, a highly porous MOF, possesses densely decorated NH<sub>2</sub> sites within its framework, resulting in a strong interaction with  $C_3H_4$ . This interaction enables a remarkable uptake capacity of up to 84.5 cm<sup>3</sup> g<sup>-1</sup> at 1 kPa and 298 K, which is almost two times higher than that of NTU-100-NO<sub>2</sub>, the same framework with decorated NO<sub>2</sub>.<sup>84</sup> Furthermore, NTU-100-NH<sub>2</sub> exhibited significantly improved adsorptive selectivity (1.4 to 11.3) and diffusion selectivity (0.13 to 3.15). Therefore, NTU-100-NH<sub>2</sub> has been shown to possess the capacity to remove trace amounts of  $C_3H_4$  from  $C_3H_4/C_3H_6$  (1/99, v/v) mixtures at room temperature, while also being capable of producing polymer-grade  $C_3H_6$  directly. Further breakthrough experiments revealed that the productivity of pure  $C_3H_6$  remained almost unchanged at different sweep rates (1, 2, and 4 mL min<sup>-1</sup>) even with wet feed gas (Fig. 20b).

The phenomenon of co-adsorption has been frequently observed in the context of adsorption separation mechanisms that are reliant upon the interaction between the host and guest molecules. NTU-88, the framework with rotational pyridyl rings that was reported by us, exhibited sieving separation of  $C_3H_4/C_3H_6$ .<sup>69</sup> At 298 K, a two-step  $C_3H_4$  adsorption process, with a steep increase occurring at 15.6 kPa and a maximum uptake of

86.0 cm<sup>3</sup> g<sup>-1</sup> was observed. Furthermore, the shutter rotating pressure was observed to shift to a very low value of 2.9 kPa at 273 K. However, the total uptake remained constant within the temperate range of 273 to 328 K. In contrast, negligible  $C_3H_6$  uptake was observed in NTU-88 within these wide temperature ranges. The observation of nearly the same elution times for  $C_3H_4/He$  and  $C_3H_4/C_3H_6$  in the dynamic separation process indicates that the co-adsorption of  $C_3H_4$  and  $C_3H_6$  was strongly restricted in NTU-88. Furthermore, breakthrough experiments demonstrated that NTU-88 has excellent ability for selective capture of  $C_3H_4$  from  $C_3H_4/C_3H_6$  mixtures, yielding highly-pure  $C_3H_6$  with a productivity of 4.10 mmol g<sup>-1</sup> at 273 K. Notably, this MOF can be scale-up synthesized *via* room temperature stirring of the related reagents in methanol in a very short time (Fig. 20c).

NKMOF-1-M (M = Cu or Ni) displayed the preferred adsorption of trace  $C_3H_4$  from  $C_3H_6$ . Both MOFs demonstrated high adsorption of  $C_3H_4$ , including  $CH_3C\equiv CH$  and  $CH_2=C=CH_2$  (over 3.0 mmol g<sup>-1</sup> at 298 K, 1 bar), which is considerably higher than that of  $C_3H_6$  (over 1.8 mmol g<sup>-1</sup>). Importantly, the steep curves observed at pressures below 1 kPa, across a wide temperature range, indicated strong binding affinity for  $C_3H_4$ . Therefore, their adsorption selectivity is very high, over 100. Breakthrough experiments showed that NKMOF-1-Ni can selectively capture trace amounts of  $C_3H_4$ , yielding highly pure  $C_3H_6$  (99.996%, 230 mmol L<sup>-1</sup>) from the feed gas of  $CH_3C\equiv CH/CH_2=C=CH_2/C_3H_6$  (0.5/0.5/99) mixtures at room temperature. A similar phenomenon has been observed in ultra-microporous MOF-NKMOF-M (Fig. 20d).<sup>52</sup>

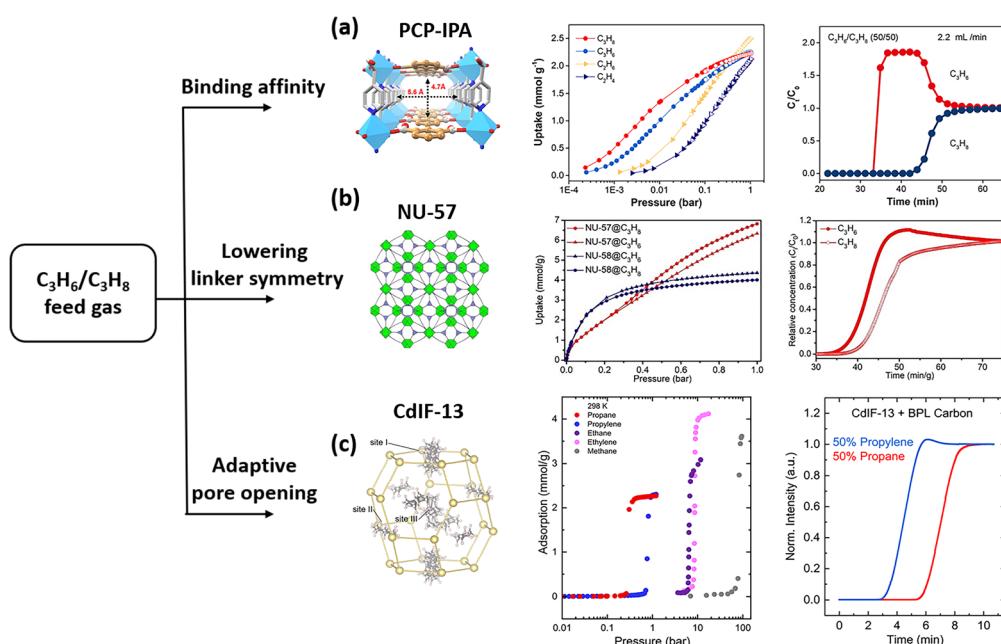


Fig. 21 (a) The 3D structure of PCP-IPA, the adsorption isotherms of different light hydrocarbons at 298 K, and breakthrough experimental curves of equimolar  $C_3H_6/C_3H_8$  mixtures at 298 K and 1 bar. (b) The structure of NU-57, the adsorption isotherms of  $C_3H_6$  and  $C_3H_8$  for NU-57 and NU-58, and breakthrough experimental curves of equimolar  $C_3H_6/C_3H_8$  mixtures at 298 K. (c) The crystal structure of  $C_3H_8$  loaded CdIF-13, adsorption isotherms for  $C_3H_6$  and  $C_3H_8$  at 298 K, and breakthrough experimental curves of equimolar  $C_3H_6/C_3H_8$  mixtures at 298 K and 1 bar. Reprinted with permission from ref. 129. Copyright 2022 Springer Nature Limited, ref. 130. Copyright 2025 John Wiley & Sons, ref. 43. Copyright 2023 American Chemical Society, respectively.



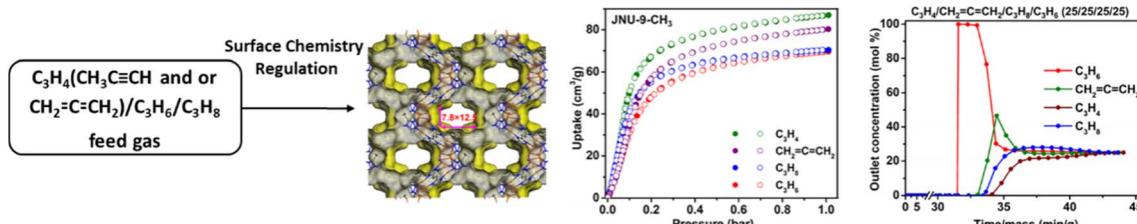


Fig. 22 The crystal structure of JNU-9-CH<sub>3</sub>, adsorption isotherms for C<sub>3</sub>H<sub>4</sub>, CH<sub>2</sub>=C=CH<sub>2</sub>, C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> at 298 K, and breakthrough experimental curves of C<sub>3</sub>H<sub>4</sub>, CH<sub>2</sub>=C=CH<sub>2</sub>, C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> (25/25/25/25, v/v) mixtures at 298 K. Reprinted with permission from ref. 60. Copyright 2024 American Chemical Society.

**3.4.2 C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub> feed gas.** Selective alkane trapping has been considered as an energy-saving method for olefin purification. Featuring parallelly aligned and extended isophthalic acids, serving as a paraffin nano-trap, PCP-IPA has been investigated for selective C<sub>3</sub>H<sub>8</sub> capture for harvesting high-purity C<sub>3</sub>H<sub>6</sub> directly.<sup>129</sup> Due to the rigidity of the framework, type-I C<sub>3</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> adsorption isotherms were observed. C<sub>3</sub>H<sub>8</sub> uptake was always higher than that of C<sub>3</sub>H<sub>6</sub> in the pressure range of 0–1 bar. Thereby, the calculated selectivity of C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> (1/1, v/v) was determined to be 2.48, at 298 K and 1 bar, including a separation potential of 1.2 mol L<sup>-1</sup>. Breakthrough results (C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub>: 1/1, v/v) confirmed the separation ability under dynamic conditions, that is highly-pure C<sub>3</sub>H<sub>6</sub> broke the column at 34.8 min, while C<sub>3</sub>H<sub>8</sub> eluted out at 43.9 min. The productivity of C<sub>3</sub>H<sub>6</sub> (99.99%) was 15.23 L kg<sup>-1</sup> (Fig. 21a).

A desymmetrization approach was utilised in the preparation of NU-57, which realized inversion C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> separation. The maximum C<sub>3</sub>H<sub>8</sub> uptake of NU-57 was found to be superior to that of C<sub>3</sub>H<sub>6</sub>, despite the adsorption isotherms of C<sub>3</sub>H<sub>8</sub> and C<sub>3</sub>H<sub>6</sub> demonstrating almost complete overlap in the low-pressure region.<sup>130</sup> In addition, the Q<sub>st</sub> values of NU-57 for C<sub>3</sub>H<sub>8</sub> and C<sub>3</sub>H<sub>6</sub> were 33.0 kJ mol<sup>-1</sup> and 28.5 kJ mol<sup>-1</sup>, respectively, suggesting a stronger interaction between the framework and C<sub>3</sub>H<sub>8</sub> molecules than C<sub>3</sub>H<sub>6</sub> molecules. Harvesting of pure C<sub>3</sub>H<sub>6</sub> from C<sub>3</sub>H<sub>8</sub>-containing mixtures was confirmed by breakthrough experiments. A similar approach was adopted in Cu-ASY,<sup>131</sup> a new MOF that incorporates infinite Cu-carboxylate rods which can separate C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> mixtures with a C<sub>3</sub>H<sub>6</sub> productivity of 2.2 L kg<sup>-1</sup> (Fig. 21b).

In addition to the aforementioned rigid frameworks, which have been demonstrated to be capable of performing inversion C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> separation, a soft framework has also been shown to be effective for this task. CdIF-13, prepared from Cd<sup>2+</sup> and benzimidazolate, exhibited guest-induced structural changes.<sup>43</sup> At 298 K, step-shaped C<sub>3</sub>H<sub>8</sub> adsorption isotherms revealed that gate opening of CdIF-13 occurred at 0.3 bar, which is far earlier than that of C<sub>3</sub>H<sub>6</sub> (0.8 bar). Comparatively, the isostructural ZIF-7 displayed a very small difference in gate-opening pressure of C<sub>3</sub>H<sub>8</sub> (0.008 bar) and C<sub>3</sub>H<sub>6</sub> (0.020 bar) at 298 K. Therefore, highly pure C<sub>3</sub>H<sub>6</sub> was harvested by using a double column packed with CdIF-13 and BPL carbon from equimolar C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> feed gas (Fig. 21c).

**3.4.3 C<sub>3</sub>H<sub>4</sub>(CH<sub>3</sub>C≡CH and or CH<sub>2</sub>=C=CH<sub>2</sub>)/C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub> feed gas.** The purification of C<sub>3</sub>H<sub>6</sub> from four-component

mixtures (C<sub>3</sub>H<sub>4</sub>/CH<sub>2</sub>=C=CH<sub>2</sub>/C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub>) represents the most challenging process in light hydrocarbon separations. JNU-9-CH<sub>3</sub>, a hydrolytically stable Cu<sub>4</sub>I<sub>4</sub>-triazolate MOF with a spatial distribution of polar and nonpolar sites, was designed.<sup>60</sup> The densely packed and readily accessible iodine and nitrogen sites in the 1D channel allowed JNU-9-CH<sub>3</sub> to show the single gas uptakes in a sequence of C<sub>3</sub>H<sub>8</sub> > CH<sub>3</sub>C≡CH > CH<sub>2</sub>=C=CH<sub>2</sub> > C<sub>3</sub>H<sub>6</sub>. The calculated selectivity of the C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> (1/1, v/v) mixture is 1.5 at 298 K and 1 bar, while the selectivities of CH<sub>3</sub>C≡CH/C<sub>3</sub>H<sub>6</sub> (1/1, v/v) and CH<sub>2</sub>=C=CH<sub>2</sub>/C<sub>3</sub>H<sub>6</sub> (1/1, v/v) are 2.1 and 1.3, respectively. Breakthrough experiments showed that JNU-9-CH<sub>3</sub> is able to separate such equimolar quaternary hydrocarbons, yielding highly pure C<sub>3</sub>H<sub>6</sub> of 102.7 mmol L<sup>-1</sup> (Fig. 22). In addition, the velocity of the feed gas exerts minimal influence on the separation performance.

## 4. Large-scale synthesis

Since their first appearance, MOFs have undergone remarkable development and have demonstrated excellent performance in the field of gas separation. If MOFs with gas separation functions can be synthesized on a large scale, it will undoubtedly lead to a significant reduction in production costs and bring significant benefits for the application of MOFs in the market. Despite considerable efforts, only a few of the MOF products with unique structures and potential applications have been synthesized on a large scale. The synthesis of MOFs on a large scale is still challenging due to the high cost of some of the organic ligands and metal salts that are essential components in the synthesis of MOFs. Ensuring consistent quality and reproducibility, a key requirement for large-scale production, remains a formidable task. It is therefore widely recognised in the scientific community that the large-scale synthesis of MOFs should focus on the following key aspects. First, the material cost must be reduced, which can be achieved by exploring alternative raw materials or more efficient synthesis routes. Second, higher yields should be achieved to meet the growing demands of various industries. Thirdly, a relatively fast and simple synthesis process is highly desirable, which can increase production efficiency and reduce production time. Fourthly, the synthesis process should have less impact on the environment, in line with the principles of sustainable development.<sup>142,143</sup>

In addressing the complex issues hindering large-scale synthesis of MOFs, researchers have developed a range of



Table 6 Advantages and drawbacks of common synthesis methods

Methods	Pros.	Cons.
Room temperature stirring	Green, simple, and low energy	Not universally applicable for MOF preparation
Mechanochemistry	Environmental friendly, simple, and low-cost	Damaged material and high impurity content
Solvothermal	Simple	Long-reaction-cycle
Flow chemistry	High-yield production with excellent batch consistency	High cost of the reactor
Electrochemistry	High purity	Clog-prone piping
		A relatively high cost

methods, including room-temperature stirring, solvothermal, mechanochemical synthesis, continuous flow production and electrochemical which have been explored with the aim of improving production efficiency.<sup>144–147</sup> These large-scale synthesis methods have their own advantages and shortcomings, which are detailed in Table 6. The employment of such methodologies has yielded encouraging outcomes with regard to enhancing the efficiency of the production process. In the context of MOF production, it is imperative to give due consideration to both extrinsic and intrinsic factors. Poor management of these factors can lead to increased costs in the production process, which is a significant concern for both academic research and industrial applications. Despite the difficulties in large-scale production of MOFs, companies such as BASF, Framergy, Nuada or MOF apps are now able to produce a range of MOFs. MIL-100 (Fe), UiO-66 (Zr), Al-fum MOF, ZIF-8, ZIF-67, HKUST-1, PCN-250 (Fe) or MIL-127 (Fe) can be produced on a kilogram scale.<sup>148</sup> CALF-20, a MOF material that can efficiently capture CO<sub>2</sub>, in the presence of moisture after

combustion,<sup>149</sup> is now being produced by BASF on a multi-ton scale. The success of large-scale synthesis of these examples has shed light on the large-scale synthesis of MOFs (Fig. 23).

#### 4.1 Room-temperature stirring

As an alternative fast and simple method, room-temperature stirring has been a well-established for material synthesis in laboratory and industry, due to its significant advantages of cost-effectiveness, safety, scalability, versatility, simple operation, and almost negligible energy consumption. For MOF synthesis, the main procedure involves the dissolution of suitable organic ligands and metal salts separately in appropriate solvents, followed by the gradual pouring of one of the above solutions into another under stirring conditions, or the simultaneous pouring of both solutions into a third vessel. The gentle nature of this method renders it less likely to result in the rapid precipitation of reactants, thus necessitating extended periods to achieve complete crystallisation at room temperature in comparison to methods conducted at elevated temperatures. The incorporation of reaction promoters, such as formic acid, acetic acid, triethylamine, or ammonia water, has been demonstrated to effectively modulate the reaction rate. It is also crucial to ensure the homogeneity of the mixture throughout the reaction vessel to avoid the formation of undesirable particles due to localised supersaturation.

NKMOF-8-Br and NKMOF-8-Me, which possess excellent separation performance for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub>, can be efficiently prepared through room temperature stirring of ligands, CuI, and triethylamine (TEA) in acetonitrile at room temperature within 3 minutes. Both MOFs have high yields (>90%). Notably, the acetonitrile solution can be readily recycled through filtration and then reused in the following reaction cycles without the need for further purification.<sup>150</sup> We have recently achieved large-scale synthesis of a series of light hydrocarbon separators, including NTU-88 (60.0 g, stirring in 1 min) and NTU-101-NH<sub>2</sub> (50.5 g, 5 min stirring) via room temperature stirring of N containing ligands and Cu<sup>2+</sup> or Ni<sup>2+</sup>.<sup>62,69</sup> For them, ammonia or triethylamine was gently added into the stirring system. Meanwhile, a MOF of CeBTB (H<sub>3</sub>BTB: 1,3,5-Tri(4-carboxyphenyl) benzene) with a carboxylate ligand can also be synthesized by such a method.<sup>151</sup> Differently, the promoter 1-methylimidazole was used to regulate the deprotonation ability of H<sub>3</sub>BTB. CeBTB (uniform size of 150 nm) can be conveniently prepared in a 1 L beaker. As a C<sub>3</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>8</sub> separator, ZJU-75 can

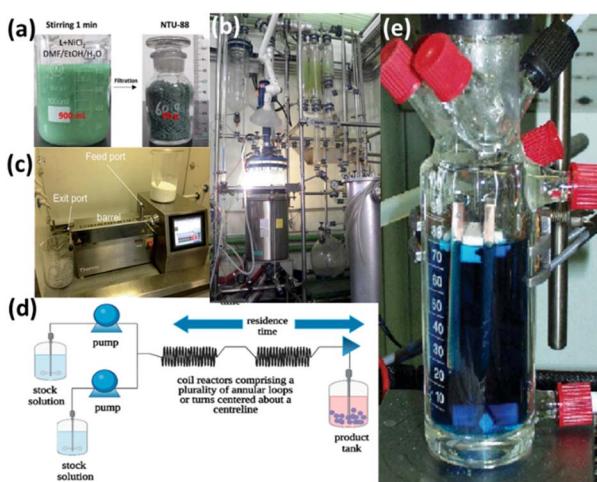


Fig. 23 (a) Large-scale preparation of NTU-88 via room temperature stirring. (b) Large-scale preparation of Uio-66 using solvothermal methods. (c) Key components of the twin-screw extruder. (d) Demonstration of the continuous flow method for synthesizing MOFs. (e) Synthesis of MOFs using Cu as the electrode material by the electrochemical synthesis method. Reprinted with permission from ref. 69. Copyright 2023 John Wiley & Sons, ref. 162. Copyright 2015 Elsevier, ref. 152. Copyright 2015 The Royal Society of Chemistry, ref. 163. Copyright 2022 The Royal Society of Chemistry, ref. 164. 2006. Copyright The Royal Society of Chemistry, respectively.



also be readily scaled up through room temperature stirring of  $K_2[Ni(CN)_4] \cdot nH_2O$  pyz-NH<sub>2</sub> and  $Co(NO_3)_2 \cdot 6H_2O$  in water/ethanol solution.<sup>90</sup> Further kilogram-scale synthesis has also been achieved in this case, giving a high yield of 75% within 8 hours. The calculated space-time yield was found to be 158 kg per m<sup>3</sup> per day. It is interesting that MOF-303 can be prepared in water as the sole solvent and can be readily scaled up to the kilogram level with a yield of up to 93%, but a reflux reaction apparatus is required.<sup>123</sup>

#### 4.2 Mechanochemical synthesis

The concept of mechanochemistry is regarded as one of the important innovations that have had a profound impact on the world.<sup>152</sup> The application of mechanical forces can break the intramolecular bonds between reactants, thereby facilitating the formation of new bonds. This process is accompanied by the transformation of mechanical energy into thermal energy through the process of heating.<sup>153</sup> The mechanochemical synthesis has been shown to be solvent-free or to require only a minimal amount of solvent, thereby minimising its environmental impact. Furthermore, mechanochemistry has been demonstrated to exhibit advantages over the above-mentioned strategies, including reduced reaction times and enhanced efficiency.<sup>154–156</sup> This characteristic is of particular significance for poorly soluble metal precursors or ligands, as it circumvents the issues associated with solubility whilst concomitantly minimizing environmental damage. By reducing the reliance on solvents, there is the potential to substantially reduce production costs, a highly desirable outcome in both academic research and industrial production.

A mixture of dried copper acetate and isonicotinic acid was subjected to 10 minutes of milling, resulting in the formation of copper(II) isonicotinic acid and the trapping of acetic acid and water molecules within the pores of the resultant product.<sup>157</sup> Notwithstanding the potential of mechanical milling for the synthesis of MOFs, the majority of extant research has focused on gram-scale feeds. In order to find ways to scale-up this technique, a ThermoFisher Process-11 Twin Screw Extruder, a machine with a screw configuration of a sequence of alternating conveying and kneading zones, was used.<sup>152</sup> Through the feed port, the MOF precursor can be introduced into a heatable drum containing screws and the exit port can be connected to a mold for collecting the final material, yielding continuous synthesis of various metal complexes, including Ni(salen),  $Ni(NCS)_2(PPh_3)_2$ , HKUST-1, ZIF-8, and MAF-4 Al(fumarate)(OH). It is important to note that the space time yields for these methods are as high as  $144 \times 10^3$  kg m<sup>-3</sup> per day, which is significantly greater than the yields for other methods of MOF synthesis. Recent breakthroughs in mechanochemical synthesis have demonstrated the remarkable efficiency of a novel “cage-on-MOF” strategy, enabling the rapid preparation of 28 distinct MOF@PCC composites within just 5 min.<sup>158</sup> Particularly noteworthy is the exceptional performance of these materials, with MOF-808@PCC-4 showing dramatic enhancements in  $CO_2/C_2H_2$  separation (exhibiting a 64% increase in  $C_2H_2$  uptake capacity and 166% improvement in IAST selectivity

in breakthrough tests), while MIL-101@PCC-4 achieved outstanding  $C_2H_2$  adsorption reaching 6.11 mmol g<sup>-1</sup> with an IAST selectivity of 2.59. This mechanochemical approach represents a significant advancement compared to conventional solution-based post-synthetic modification methods, which typically require time-consuming heating procedures and multiple purification steps. The “cage-on-MOF” methodology not only achieves gram-scale synthesis (up to 100 g batches) with near-quantitative yields, but also maintains excellent crystallinity and porosity control while completely eliminating organic solvent consumption—offering a synthesis efficiency approximately 300 times greater than that of traditional solution-phase methods while ensuring consistent material quality.

Although mechanochemistry has evident advantages, this strategy is not universally applicable as the generation of excessive mechanical energy may lead to the amorphization of MOFs and the subsequent decomposition and damage of the material.<sup>159</sup> In addition, the formation of new bonds is often inconsistent, leading to the production of various impurities that cannot be easily removed. Furthermore, mechanical synthesis is often not suitable for MOFs with high coordination numbers that require a relatively slow and well-controlled crystallization process.

#### 4.3 Solvothermal and hydrothermal synthesis

In 1995, Nalco Chemical Company and the Yaghi group produced the first MOF materials using solvothermal synthesis. A certain proportion of organic ligands and metal salts are dissolved in different solvents, after which the solutions are mixed in a sealed reaction vessel, and then the nucleation and growth of crystals is promoted by heating.<sup>160</sup> This method is by far the most widely used method for producing MOF crystals in laboratories.<sup>161</sup>

However, there are significant limitations for the large-scale production of MOFs. In particular, increasing the size of the reaction vessel leads to a significant reduction in the surface-to-volume ratio, which further reduces the efficiency of the reaction, but also requires, for example, the use of hundreds of liters of solvent per batch. In addition, the reaction time is too long, given the high temperature and pressure involved. Finally, the cost of the pressure vessel, especially those for hundreds of liters, is very high.

#### 4.4 Flow chemistry synthesis

Flow chemistry is a process in which the solutions of various components are passed through a reaction apparatus, leading to the occurrence of chemical reactions.<sup>163,165,166</sup> This process contrasts with traditional batch-reaction devices, in which different components are placed in separate vessels and react over time. Consequently, flow chemistry can be regulated precisely through parameter change and device design, thereby ensuring reproducibility of high-quality products. In the context of flow chemistry, the most salient benefit of this process is that the rapid separation of MOF crystals and related solvents facilitates the recycling of solvents. This, in turn, has a number



of significant benefits, such as environmental friendliness, capital investment and energy input (hot solvent will be transferred into the reactor in time).

The utilisation of both a stirred tank reactor and a plug flow reactor has become a widespread practice in the field of flow synthesis. For instance, a microwave-assisted flow chemistry method has been employed to synthesize  $\text{UiO-66}$  with a significantly enhanced reaction rate when compared to conventional solvothermal synthesis.<sup>162</sup> In addition,  $\text{MOF-5}$  with a space time yield of 0.1 ton per  $\text{m}^3$  per day, has been synthesised *via* a flow chemistry method.<sup>167</sup> After optimization, the flow system, operated at 140 °C and residence time  $\geq 5.5$  h, reached a steady state that can continuously produce high surface area  $\text{MOF-5}$ . In other words, parameter optimization in flow chemistry plays an important role in the manufacture of MOFs with controlled quality.

Following the modification of the heating reactor to incorporate a microwave-assisted heating reactor, a notable enhancement in the rate of MOF growth, accompanied by excellent reproducibility, has been observed in the case of  $\text{MOF-74(Ni)}$ .<sup>168</sup> The raw materials of  $\text{MOF-74(Ni)}$  were introduced into a microwave reactor to facilitate initial nucleation. The solution was subsequently introduced into a reactor for a duration of 8 minutes to complete the growth of the crystals. This process resulted in enhanced crystallinity, accompanied by a space time yield of 2160 kg per  $\text{m}^3$  per day. Using a similar system,  $\text{UiO-66}$ ,  $\text{HKUST-1}$  and  $\text{MIL-53(Al)}$  were obtained in 7, 1 and 4 minutes of residence time, resulting in high space time yields of 7204 kg per  $\text{m}^3$  per day, 64 800 kg per  $\text{m}^3$  per day and 3618 kg per  $\text{m}^3$  per day, respectively.<sup>146</sup> In such a system, the separation of the nucleation and growth steps has been shown to accelerate the reaction rate and ensure the quality of the resulting MOFs.

However, it was also noted that the flow chemistry synthesis suffers from a problem of reactor/pipeline fouling and clogging. During the reaction process, precipitates are often generated, and the clogging caused by these precipitates can accumulate in the system. As a consequence, the efficiency and quality of the product are affected. In order to address this issue, it is imperative to implement regular cleaning procedures or to undertake a redesign of the reactor components, such as the introduction of a mesh screw inside the tube, in order to maintain the functionality of the flow system.

#### 4.5 Electrochemical synthesis

In comparison with the conventional solvothermal synthesis method, the electrochemical synthesis has milder reaction conditions, a significantly reduced reaction time, and the process of MOF synthesis can be regulated by modulating the voltage.<sup>148</sup> With regard to electrochemical synthesis, the methods can be categorised into three distinct classifications depending on the direction of electron transfer and the location where the nucleation process occurs: (1) surface electrode nucleation, including anodic dissolution and cathodic deposition, (2) indirect bipolar electrodeposition and (3) electrophoretic deposition.<sup>169</sup> The researchers in PASF immersed a copper plate as an anode into an organic solution containing 1,3,5-

benzenetricarboxylic acid. When a specific current was applied, Cu ions were transferred from the surface of the anode to the solution and reacted with the organic linker in the solution, ultimately producing an octahedral powder of  $\text{HKUST-1}$ . Notably, the surface area of the electrochemical synthesis material is  $1820 \text{ m}^2 \text{ g}^{-1}$ , which is larger than that of  $\text{HKUST-1}$  synthesised by the solvothermal method ( $1550 \text{ m}^2 \text{ g}^{-1}$ ). Recently, a mechanism for the anodic solvation synthesis of  $\text{HKUST-1}$  was proposed, which consists of initial nucleation; growth of  $\text{HKUST-1}$  islands; intergrowth; and crystal detachment.<sup>170</sup>

Despite the advances achieved thus far, numerous challenges have been encountered when attempting to realise large-scale production. Initially, a large-scale electrochemical synthesis device is required; however, this task involves complex issues. For example, the electrodes must be uniformly arranged over a large area, and the circulation and stirring of the electrolyte must be well-controlled, which pose significant technical difficulties. Furthermore, when reaction control is considered during large-scale synthesis, ensuring the homogeneity of parameters such as electric field, temperature, and electrolyte concentration throughout the entire reaction system is extremely challenging, as doing so would produce products of unstable quality and poor consistency.

In a word, room-temperature stirring is one of the most practical and effective methods for large-scale synthesis of MOFs, especially holding a unique position in the large-scale synthesis of MOFs for light hydrocarbon separation. Secondly, mechanochemical methods have emerged and shown potential in the large-scale synthesis of light hydrocarbon separation MOFs. Although solvothermal, electrochemical and flow chemistry methods have made some progress in the large-scale synthesis of some MOFs, they still need to be studied more in the large-scale synthesis of light hydrocarbon separation MOFs. It is believed that in the near future, in terms of light hydrocarbon separation, the large-scale production methods are more diversified, the industrialization conditions will be more mature, and the large-scale production of high-efficiency, low-cost and sustainable high-quality light hydrocarbon separation MOFs will be realized. It is encouraging to see that companies like BASF are already striving for excellence and have achieved exciting results in the large-scale production of MOFs. It is encouraging that companies like BASF are already pursuing excellence and achieving exciting results in the large-scale production of MOFs. It is expected that this trend will continue, leading to significant progress in the large-scale production and application of MOFs not only in light hydrocarbon separation but even in other areas.

## 5. Conclusion and outlook

The development of high-performance MOFs for light hydrocarbon separation has witnessed remarkable progress, driven by innovative design strategies and engineering advancements. Rigid MOFs, functionalized with tailored binding sites, enable selective guest interactions through precise control of pore environments, while flexible MOFs leverage dynamic structural



responses under external stimuli to achieve molecular recognition. These advancements, coupled with emerging scale-up synthesis techniques, underscore the potential of MOFs to revolutionize industrial separation processes. However, critical challenges remain in achieving cost-effective, one-step separation processes, precise material design for specific applications, and scalable production. Addressing these challenges will require interdisciplinary collaboration and sustained innovation in both fundamental research and applied technologies.

To propel MOFs toward practical implementation in light hydrocarbon separation, future research should prioritize the following directions.

### 5.1 Pore aperture engineering

Precise control over pore size and geometry is essential for molecular sieving. While rigid MOFs allow tunability *via* ligand and metal node selection, flexible MOFs offer dynamic pore adjustment in response to guest molecules. Integrating computational tools—such as artificial intelligence (AI) and molecular simulations—will accelerate the prediction of optimal MOF architectures for target separations. These approaches can minimize trial-and-error experimentation and guide the design of frameworks with pore environments tailored to specific gas pairs.

### 5.2 Enhanced host–guest interactions

Synergistic integration of multiple interaction mechanisms (*e.g.*, van der Waals forces, open metal sites, hydrogen bonding, and  $\pi$ -complexation) within a single MOF can amplify selectivity. Future work should focus on multifunctional frameworks that combine complementary binding sites, such as Lewis acidic metal clusters and functionalized organic linkers, to enable multi-mechanism adsorption. Additionally, exploring stimuli-responsive systems (*e.g.*, light- or temperature-triggered structural changes) could further refine dynamic separation processes.

### 5.3 Elucidating mechanisms

A deeper understanding of the interplay between thermodynamic equilibria and kinetic diffusion is critical. Advanced characterization techniques (*e.g.*, *in situ* spectroscopy and *operando* diffraction) combined with molecular dynamics (MD) simulations can unravel molecular transport pathways and adsorption site competition. Investigating the nanocapnification effects within MOF pores may also reveal strategies to manipulate guest molecule behavior, particularly under industrially relevant high-temperature conditions.

### 5.4 Scalability and stability

Bridging the gap between laboratory synthesis and industrial deployment demands scalable, low-cost production methods. Room temperature stirring synthesis and solvent-free routes should be optimized to reduce energy consumption and material waste. Concurrently, improving MOF stability under harsh

operational conditions (*e.g.*, moisture, high pressure, and thermal cycling) is imperative for long-term viability.

### 5.5 Interdisciplinary collaboration

Cross-disciplinary efforts integrating chemistry, materials science, chemical engineering, and data science will be vital to overcome existing barriers. Partnerships with industry can accelerate technology transfer, while open-access databases for MOF properties and performance metrics will streamline material discovery.

In conclusion, MOFs represent a transformative platform for light hydrocarbon separation, offering unprecedented control over molecular recognition and process efficiency. While challenges in design precision, stability, and scalability persist, the convergence of computational modeling, advanced synthesis, and mechanistic insights holds immense promise. By addressing these hurdles through targeted research and collaborative innovation, MOFs may soon transition from academic breakthroughs to industrial mainstays, ushering in a new era of sustainable and energy-efficient chemical separations.

## Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this review.

## Author contributions

J. D. conceived the idea of this project. H. Z., J. T. and C. Y. collected literatures, wrote and revised the paper. All authors offered constructive comments on the manuscript.

## Conflicts of interest

The authors declare no conflicts of interest.

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## References

- 1 D. S. Sholl and R. P. Lively, *Nature*, 2016, **532**, 435–437.
- 2 W. D. Fan, X. Wang, X. R. Zhang, X. P. Liu, Y. T. Wang, Z. X. Kang, F. N. Dai, B. Xu, R. M. Wang and D. F. Sun, *ACS Cent. Sci.*, 2019, **5**, 1261–1268.
- 3 G.-D. Wang, Y.-Z. Li, W.-J. Shi, L. Hou, Y.-Y. Wang and Z. Zhu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202311654.



4 S. A. Chernyak, M. Corda, J. P. Dath, V. V. Ordomsky and A. Y. Khodakov, *Chem. Soc. Rev.*, 2022, **51**, 7994–8044.

5 P. Hu, J. L. Hu, H. Liu, H. Wang, J. Zhou, R. Krishna and H. B. Ji, *ACS Cent. Sci.*, 2022, **8**, 1159–1168.

6 H. Wang, X. Dong, V. Colombo, Q. Wang, Y. Liu, W. Liu, X.-L. Wang, X.-Y. Huang, D. M. Proserpio, A. Sironi, Y. Han and J. Li, *Adv. Mater.*, 2018, **30**, 1805088.

7 J. Q. Liu, Y. J. Liu, D. K. Talay, E. Calverley, M. Brayden and M. Martinez, *Carbon*, 2015, **85**, 201–211.

8 Q. Z. Xue, X. F. Li, X. Chang, C. C. Ling, L. Zhu and W. Xing, *Appl. Surf. Sci.*, 2018, **444**, 772–779.

9 Y. F. Yuan, Y. S. Wang, X. L. Zhang, W. C. Li, G. P. Hao, L. Han and A. H. Lu, *Angew. Chem., Int. Ed.*, 2021, **60**, 19063–19067.

10 Y. Chai, X. Han, W. Li, S. Liu, S. Yao, C. Wang, W. Shi, I. da-Silva, P. Manuel, Y. Cheng, L. D. Daemen, A. J. Ramirez-Cuesta, C. C. Tang, L. Jiang, S. Yang, N. Guan and L. Li, *Science*, 2020, **368**, 1002–1006.

11 J. G. Min, K. C. Kemp and S. B. Hong, *Sep. Purif. Technol.*, 2020, **250**, 117146.

12 H. Li, M. Eddaoudi, M. O'Keeffe and O. M. Yaghi, *Nature*, 1999, **402**, 276–279.

13 X. Wang, W. Lu, Z. Y. Gu, Z. Wei and H. C. Zhou, *Chem. Commun.*, 2016, **52**, 1926–1929.

14 K. S. Walton, *Nat. Chem.*, 2014, **6**, 277–278.

15 S. Krause, V. Bon, I. Senkovska, U. Stoeck, D. Wallacher, D. M. Tobbens, S. Zander, R. S. Pillai, G. Maurin, F. X. Coudert and S. Kaskel, *Nature*, 2016, **532**, 348–352.

16 S. H. Yang, X. Lin, W. Lewis, M. Suyetin, E. Bichoutskaia, J. E. Parker, C. C. Tang, D. R. Allan, P. J. Rizkallah, P. Hubberstey, N. R. Champness, K. M. Thomas, A. J. Blake and M. Schröder, *Nat. Mater.*, 2012, **11**, 710–716.

17 L. Wen, P. Cheng and W. B. Lin, *Chem. Sci.*, 2012, **3**, 2288–2292.

18 S. Kitagawa, R. Kitaura and S. Noro, *Angew. Chem., Int. Ed.*, 2004, **43**, 2334–2375.

19 H. Sato, W. Kosaka, R. Matsuda, A. Hori, Y. Hijikata, R. V. Belosludov, S. Sakaki, M. Takata and S. Kitagawa, *Science*, 2014, **343**, 167–170.

20 J. Duan, M. Higuchi, R. Krishna, T. Kiyonaga, Y. Tsutsumi, Y. Sato, Y. Kubota, M. Takata and S. Kitagawa, *Chem. Sci.*, 2014, **5**, 660–666.

21 Z.-J. Liang, F.-D. Dong, L. Ye, K. Zheng, D.-Y. Hu, X. Feng, W.-Y. Su, Z.-S. Wang, M.-Y. Zhou, Z.-L. Fang, D.-D. Zhou, J.-P. Zhang and X.-M. Chen, *Chem. Sci.*, 2025, **16**, 3307–3312.

22 Y. S. Bae, C. Y. Lee, K. C. Kim, O. K. Farha, P. Nickias, J. T. Hupp, S. T. Nguyen and R. Q. Snurr, *Angew. Chem., Int. Ed.*, 2012, **51**, 1857–1860.

23 Y.-L. Peng, T. Wang, C. Jin, C.-H. Deng, Y. Zhao, W. Liu, K. A. Forrest, R. Krishna, Y. Chen, T. Pham, B. Space, P. Cheng, M. J. Zaworotko and Z. Zhang, *Nat. Commun.*, 2021, **12**, 5768.

24 X. W. Gu, J. X. Wang, E. Y. Wu, H. Wu, W. Zhou, G. D. Qian, B. L. Chen and B. Li, *J. Am. Chem. Soc.*, 2022, **144**, 2614–2623.

25 Z. Z. Lu, H. G. W. Godfrey, I. da Silva, Y. Q. Cheng, M. Savage, P. Manuel, S. Rudic, A. J. Ramirez-Cuesta, S. H. Yang and M. Schroder, *Chem. Sci.*, 2018, **9**, 3401–3408.

26 S. Horike, Y. Inubushi, T. Hori, T. Fukushima and S. Kitagawa, *Chem. Sci.*, 2012, **3**, 116–120.

27 X. Huang, F. Chen, H. Sun, L. Yang, Q. Yang, Z. Zhang, Y. Yang, Q. Ren and Z. Bao, *J. Am. Chem. Soc.*, 2023, **146**, 617–626.

28 C.-X. Chen, Z.-W. Wei, T. Pham, P. C. Lan, L. Zhang, K. A. Forrest, S. Chen, A. M. Al-Enizi, A. Nafady, C.-Y. Su and S. Ma, *Angew. Chem., Int. Ed.*, 2021, **60**, 9680–9685.

29 H. Angewandte ChemieChen, B. Wang, B. Zhang, J. Chen, J. Gui, X. Shi, W. Yan, J. Li and L. Li, *Chem. Sci.*, 2023, **14**, 7068–7075.

30 Q. Min Wang, D. Shen, M. Bülow, M. Ling Lau, S. Deng, F. R. Fitch, N. O. Lemcoff and J. Semanscin, *Microporous Mesoporous Mater.*, 2002, **55**, 217–230.

31 E. D. Bloch, W. L. Queen, R. Krishna, J. M. Zadrozny, C. M. Brown and J. R. Long, *Science*, 2012, **335**, 1606–1610.

32 A. Cadiou, K. Adil, P. M. Bhatt, Y. Belmabkhout and M. Eddaoudi, *Science*, 2016, **353**, 137–140.

33 J. Cui, Z. Zhang, L. Yang, J. Hu, A. Jin, Z. Yang, Y. Zhao, B. Meng, Y. Zhou, J. Wang, Y. Su, J. Wang, X. Cui and H. Xing, *Science*, 2024, **383**, 179–183.

34 X. Cui, K. Chen, H. Xing, Q. Yang, R. Krishna, Z. Bao, H. Wu, W. Zhou, X. Dong, Y. Han, B. Li, Q. Ren, M. J. Zaworotko and B. Chen, *Science*, 2016, **353**, 141–144.

35 Q. Ding, Z. Zhang, C. Yu, P. Zhang, J. Wang, X. Cui, C.-H. He, S. Deng and H. Xing, *Sci. Adv.*, 2020, **6**, eaaz4322.

36 Q. Dong, Y. Huang, K. Hyeon-Deuk, I. Y. Chang, J. Wan, C. Chen, J. Duan, W. Jin and S. Kitagawa, *Adv. Funct. Mater.*, 2022, **32**, 2203745.

37 Q. Dong, Y. Huang, J. Wan, Z. Lu, Z. Wang, C. Gu, J. Duan and J. Bai, *J. Am. Chem. Soc.*, 2023, **145**, 8043–8051.

38 C. Güçüyener, J. van den Bergh, J. Gascon and F. Kapteijn, *J. Am. Chem. Soc.*, 2010, **132**, 17704–17706.

39 A. N. Hong, H. Yang, T. Li, Y. Wang, Y. Wang, X. Jia, A. Zhou, E. Kusumoputro, J. Li, X. Bu and P. Feng, *ACS Appl. Mater. Interfaces*, 2021, **13**, 52160–52166.

40 Y. Huang, J. Wan, T. Pan, K. Ge, Y. Guo, J. Duan, J. Bai, W. Jin and S. Kitagawa, *J. Am. Chem. Soc.*, 2023, **145**, 24425–24432.

41 Y. Jiang, J. Hu, L. Wang, W. Sun, N. Xu, R. Krishna, S. Duttwyler, X. Cui, H. Xing and Y. Zhang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202200947.

42 J. Luo, G. Yang, G. Zhang, Z. Huang, J. Peng, Y. Luo, X. Wang, C. Yang, J. Jiang, D. Cao, B. Chen, H. Tang and J. Xiao, *Chem*, 2024, **10**, 3148–3158.

43 R. A. Klein, L. W. Bingel, A. Halder, M. Carter, B. A. Trump, E. D. Bloch, W. Zhou, K. S. Walton, C. M. Brown and C. M. McGuirk, *J. Am. Chem. Soc.*, 2023, **145**, 21955–21965.

44 B. Li, X. Cui, D. O'Nolan, H.-M. Wen, M. Jiang, R. Krishna, H. Wu, R.-B. Lin, Y.-S. Chen, D. Yuan, H. Xing, W. Zhou, Q. Ren, G. Qian, M. J. Zaworotko and B. Chen, *Adv. Mater.*, 2017, **29**, 1704210.

45 K. Li, D. H. Olson, J. Seidel, T. J. Emge, H. Gong, H. Zeng and J. Li, *J. Am. Chem. Soc.*, 2009, **131**, 10368–10369.





46 L. Li, R.-B. Lin, R. Krishna, H. Li, S. Xiang, H. Wu, J. Li, W. Zhou and B. Chen, *Science*, 2018, **362**, 443–446.

47 L. Li, R.-B. Lin, R. Krishna, X. Wang, B. Li, H. Wu, J. Li, W. Zhou and B. Chen, *J. Am. Chem. Soc.*, 2017, **139**, 7733–7736.

48 Y. Li, Y. Wu, J. Zhao, J. Duan and W. Jin, *Chem. Sci.*, 2024, **15**, 9318–9324.

49 P.-Q. Liao, W.-X. Zhang, J.-P. Zhang and X.-M. Chen, *Nat. Commun.*, 2015, **6**, 8697.

50 C. X. Zhao, X. Liu, J. N. Liu, J. Wang, X. Wan, C. Wang, X. Y. Li, J. Shui, L. Song, H. J. Peng, B. Q. Li and Q. Zhang, *Angew. Chem., Int. Ed.*, 2023, **135**, e202313028.

51 S. Liu, Y. Huang, J. Wan, J.-J. Zheng, R. Krishna, Y. Li, K. Ge, J. Tang and J. Duan, *Chem. Sci.*, 2024, **15**, 6583–6588.

52 Y. L. Peng, C. He, T. Pham, T. Wang, P. Li, R. Krishna, K. A. Forrest, A. Hogan, S. Suepaul, B. Space, M. Fang, Y. Chen, M. J. Zaworotko, J. Li, L. Li, Z. Zhang, P. Cheng and B. Chen, *Angew. Chem., Int. Ed.*, 2019, **58**, 10209–10214.

53 C. Song, F. Zheng, Y. Liu, Q. Yang, Z. Zhang, Q. Ren and Z. Bao, *Angew. Chem., Int. Ed.*, 2023, **62**, e202313855.

54 J. Cui, Z. Zhang, L. Yang, J. Hu, A. Jin, Z. Yang, Y. Zhao, B. Meng, Y. Zhou, J. Wang, Y. Su, J. Wang, X. Cui and H. Xing, *Science*, 2024, **383**, 179–183.

55 Y.-J. Tian, C. Deng, L. Zhao, J.-S. Zou, X.-C. Wu, Y. Jia, Z.-Y. Zhang, J. Zhang, Y.-L. Peng, G. Chen and M. J. Zaworotko, *Nat. Chem.*, 2025, **17**, 141–147.

56 S.-M. Wang, M. Shivanna, S.-T. Zheng, T. Pham, K. A. Forrest, Q.-Y. Yang, Q. Guan, B. Space, S. Kitagawa and M. J. Zaworotko, *J. Am. Chem. Soc.*, 2024, **146**, 4153–4161.

57 Y. Jiang, J. Hu, L. Wang, W. Sun, N. Xu, R. Krishna, S. Duttwyler, X. Cui, H. Xing and Y. Zhang, *Angew. Chem., Int. Ed.*, 2022, **134**, e202200947.

58 Y. Hu, Y. Jiang, J. Li, L. Wang, M. Steiner, R. F. Neumann, B. Luan and Y. Zhang, *Adv. Funct. Mater.*, 2023, **33**, 2213915.

59 S.-C. Xiang, Z. Zhang, C.-G. Zhao, K. Hong, X. Zhao, D.-R. Ding, M.-H. Xie, C.-D. Wu, M. C. Das, R. Gill, K. M. Thomas and B. Chen, *Nat. Commun.*, 2011, **2**, 204.

60 X.-J. Xie, Z.-H. Zhang, Q.-Y. Cao, Y.-L. Huang, D. Luo, H. Zeng, W. Lu and D. Li, *J. Am. Chem. Soc.*, 2024, **146**, 30155–30163.

61 H. Yang, F. Peng, D. E. Schier, S. A. Markotic, X. Zhao, A. N. Hong, Y. Wang, P. Feng and X. Bu, *Angew. Chem., Int. Ed.*, 2021, **60**, 11148–11152.

62 W. Yang, J. Wang, K. Tan, H. L. Zhou, M. Zhang, R. Krishna, J. Duan and L. Huang, *Angew. Chem., Int. Ed.*, 2025, **64**, e202425638.

63 Y. Ye, Y. Xie, Y. Shi, L. Gong, J. Phipps, A. M. Al-Enizi, A. Nafady, B. Chen and S. Ma, *Angew. Chem. Int. Ed. Engl.*, 2023, **62**, e202302564.

64 L. Yu, X. Han, H. Wang, S. Ullah, Q. Xia, W. Li, J. Li, I. Silva, P. Manuel, S. Rudić, Y. Cheng, S. Yang, T. Thonhauser and J. Li, *J. Am. Chem. Soc.*, 2021, **143**, 19300–19305.

65 H. Zeng, M. Xie, T. Wang, R.-J. Wei, X.-J. Xie, Y. Zhao, W. Lu and D. Li, *Nature*, 2021, **595**, 542–548.

66 L. Zhang, B. Yu, M. Wang, Y. Chen, Y. Wang, L.-B. Sun, Y.-B. Zhang, Z. Zhang, J. Li and L. Li, *Angew. Chem., Int. Ed.*, 2025, **64**, e202418853.

67 P. Zhang, Y. Zhong, Y. Zhang, Z. Zhu, Y. Liu, Y. Su, J. Chen, S. Chen, Z. Zeng, H. Xing, S. Deng and J. Wang, *Sci. Adv.*, 2022, **8**, eabn9231.

68 Y.-B. Wang, T.-F. Zhang, Y.-X. Lin, J.-X. Wang, H.-M. Wen, X. Zhang, G. Qian and B. Li, *J. Mater. Chem. A*, 2024, **12**, 25812–25819.

69 J. Wan, H. L. Zhou, K. Hyeon-Deuk, I. Y. Chang, Y. Huang, R. Krishna and J. Duan, *Angew. Chem., Int. Ed.*, 2023, **62**, e202316792.

70 R.-B. Lin, L. Li, H.-L. Zhou, H. Wu, C. He, S. Li, R. Krishna, J. Li, W. Zhou and B. Chen, *Nat. Mater.*, 2018, **17**, 1128–1133.

71 N. Nijem, H. Wu, P. Canepa, A. Marti, K. J. Balkus Jr., T. Thonhauser, J. Li and Y. J. Chabal, *J. Am. Chem. Soc.*, 2012, **134**, 15201–15204.

72 C. Gu, N. Hosono, J. J. Zheng, Y. Sato, S. Kusaka, S. Sakaki and S. Kitagawa, *Science*, 2019, **363**, 387–391.

73 Y. Wang, T. Li, L. Li, R.-B. Lin, X. Jia, Z. Chang, H.-M. Wen, X.-M. Chen and J. Li, *Adv. Mater.*, 2023, **35**, 2207955.

74 Y. Su, K.-I. Otake, J.-J. Zheng, P. Wang, Q. Lin, S. Kitagawa and C. Gu, *Nat. Commun.*, 2024, **15**, 2898.

75 S. Kitagawa and M. Kondo, *Bull. Chem. Soc. Jpn.*, 1998, **71**, 1739–1753.

76 S. Horike, S. Shimomura and S. Kitagawa, *Nat. Chem.*, 2009, **1**, 695–704.

77 U. Kökçam-Demir, A. Goldman, L. Esrafil, M. Gharib, A. Morsali, O. Weingart and C. Janiak, *Chem. Soc. Rev.*, 2020, **49**, 2751–2798.

78 J. N. Hall and P. Bollini, *React. Chem. Eng.*, 2019, **4**, 207–222.

79 L. Alaerts, M. Maes, M. A. van der Veen, P. A. Jacobs and D. E. De Vos, *Phys. Chem. Chem. Phys.*, 2009, **11**, 2903–2911.

80 S. H. Yang, A. J. Ramirez-Cuesta, R. Newby, V. Garcia-Sakai, P. Manuel, S. K. Callear, S. I. Campbell, C. C. Tang and M. Schroder, *Nat. Chem.*, 2015, **7**, 121–129.

81 Y. Duan, Y. Huang, C. Wang, Q. Wang, K. Ge, Z. Lu, H. Wang, J. Duan, J. Bai and W. Jin, *Chem. Sci.*, 2023, **14**, 4605–4611.

82 H. G. Hao, Y. F. Zhao, D. M. Chen, J. M. Yu, K. Tan, S. Q. Ma, Y. Chabal, Z. M. Zhang, J. M. Dou, Z. H. Xiao, G. Day, H. C. Zhou and T. B. Lu, *Angew. Chem., Int. Ed.*, 2018, **57**, 16067–16071.

83 E. Y. Wu, X. W. Gu, D. Liu, X. Zhang, H. Wu, W. Zhou, G. D. Qian and B. Li, *Nat. Commun.*, 2023, **14**, 6146.

84 Y. H. Huang, Y. F. Feng, Y. Li, K. Tan, J. Tang, J. F. Bai and J. G. Duan, *Angew. Chem., Int. Ed.*, 2024, **63**, e202403421.

85 Z. Q. Zhang, S. B. Peh, Y. X. Wang, C. J. Kang, W. D. Fan and D. Zhao, *Angew. Chem., Int. Ed.*, 2020, **59**, 18927–18932.

86 S. Noro, S. Kitagawa, M. Kondo and K. Seki, *Angew. Chem., Int. Ed.*, 2000, **39**, 2082–2084.

87 P. Nugent, Y. Belmabkhout, S. D. Burd, A. J. Cairns, R. Luebke, K. Forrest, T. Pham, S. Q. Ma, B. Space, L. Wojtas, M. Eddaoudi and M. J. Zaworotko, *Nature*, 2013, **495**, 80–84.

88 Y. Wang, N.-Y. Huang, X.-W. Zhang, H. He, R.-K. Huang, Z.-M. Ye, Y. Li, D.-D. Zhou, P.-Q. Liao, X.-M. Chen and J.-P. Zhang, *Angew. Chem., Int. Ed.*, 2019, **58**, 7692–7696.

89 G. D. Wang, Y. Z. Li, W. F. Zhang, L. Hou, Y. Y. Wang and Z. H. Zhu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 58862–58870.

90 D. Liu, J. Pei, X. Zhang, X.-W. Gu, H.-M. Wen, B. Chen, G. Qian and B. Li, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218590.

91 R. B. Lin, L. B. Li, H. Wu, H. Arman, B. Li, R. G. Lin, W. Zhou and B. L. Chen, *J. Am. Chem. Soc.*, 2017, **139**, 8022–8028.

92 Q. Dong, X. Zhang, S. Liu, R. B. Lin, Y. Guo, Y. Ma, A. Yonezu, R. Krishna, G. Liu, J. Duan, R. Matsuda, W. Jin and B. Chen, *Angew. Chem., Int. Ed.*, 2020, **59**, 22756–22762.

93 S. Bourrelly, P. L. Llewellyn, C. Serre, F. Millange, T. Loiseau and G. Férey, *J. Am. Chem. Soc.*, 2005, **127**, 13519–13521.

94 P. L. Llewellyn, P. Horcajada, G. Maurin, T. Devic, N. Rosenbach, S. Bourrelly, C. Serre, D. Vincent, S. Loera-Serna, Y. Filinchuk and G. Férey, *J. Am. Chem. Soc.*, 2009, **131**, 13002–13008.

95 J. A. Mason, J. Oktawiec, M. K. Taylor, M. R. Hudson, J. Rodriguez, J. E. Bachman, M. I. Gonzalez, A. Cervellino, A. Guagliardi, C. M. Brown, P. L. Llewellyn, N. Masciocchi and J. R. Long, *Nature*, 2015, **527**, 357–361.

96 M. K. Taylor, T. Runcevski, J. Oktawiec, M. I. Gonzalez, R. L. Siegelman, J. A. Mason, J. X. Ye, C. M. Brown and J. R. Long, *J. Am. Chem. Soc.*, 2016, **138**, 15019–15026.

97 S. Gao, C. G. Morris, Z. Z. Lu, Y. Yan, H. G. W. Godfrey, C. Murray, C. C. Tang, K. M. Thomas, S. H. Yang and M. Schröder, *Chem. Mater.*, 2016, **28**, 2331–2340.

98 L. B. Upi, R. Krishna, Y. Wang, X. Q. Wang, J. F. Yang and J. P. Li, *Eur. J. Inorg. Chem.*, 2016, **2016**, 4457–4462.

99 R. Kitaura, S. Kitagawa, Y. Kubota, T. C. Kobayashi, K. Kindo, Y. Mita, A. Matsuo, M. Kobayashi, H. C. Chang, T. C. Ozawa, M. Suzuki, M. Sakata and M. Takata, *Science*, 2002, **298**, 2358–2361.

100 R. Matsuda, R. Kitaura, S. Kitagawa, Y. Kubota, R. V. Belosludov, T. C. Kobayashi, H. Sakamoto, T. Chiba, M. Takata, Y. Kawazoe and Y. Mita, *Nature*, 2005, **436**, 238–241.

101 K. Kishida, Y. Watanabe, S. Horike, Y. Watanabe, Y. Okumura, Y. Hijikata, S. Sakaki and S. Kitagawa, *Eur. J. Inorg. Chem.*, 2014, **2014**, 2747–2752.

102 Y. W. Chen, Z. W. Qiao, D. F. Lv, C. X. Duan, X. J. Sun, H. X. Wu, R. F. Shi, Q. B. Xia and Z. Li, *Chem. Eng. J.*, 2017, **328**, 360–367.

103 H. Zeng, M. Xie, Y.-L. Huang, Y. Zhao, X.-J. Xie, J.-P. Bai, M.-Y. Wan, R. Krishna, W. Lu and D. Li, *Angew. Chem., Int. Ed.*, 2019, **58**, 8515–8519.

104 A. Schneemann, V. Bon, I. Schwedler, I. Senkovska, S. Kaskel and R. A. Fischer, *Chem. Soc. Rev.*, 2014, **43**, 6062–6096.

105 Y. Su, K. Otake, J. J. Zheng, S. Horike, S. Kitagawa and C. Gu, *Nature*, 2022, **611**, 289–294.

106 H. Zeng, X.-J. Xie, T. Wang, M. Xie, Y. Wang, R.-J. Wei, W. Lu and D. Li, *Nat. Chem. Eng.*, 2024, **1**, 108–115.

107 Q. J. Wang, J. B. Hu, L. F. Yang, Z. Q. Zhang, T. Ke, X. L. Cui and H. B. Xing, *Nat. Commun.*, 2022, **13**, 2955.

108 L. B. Li, H. M. Wen, C. H. He, R. B. Lin, R. Krishna, H. Wu, W. Zhou, J. P. Li, B. Li and B. L. Chen, *Angew. Chem., Int. Ed.*, 2018, **57**, 15183–15188.

109 W.-G. Cui, T.-L. Hu and X.-H. Bu, *Adv. Mater.*, 2020, **32**, 1806445.

110 C. Y. Chuah, H. Lee and T.-H. Bae, *Chem. Eng. J.*, 2022, **430**, 132654.

111 S. Q. Yang and T. L. Hu, *Coord. Chem. Rev.*, 2022, **468**, 214628.

112 L. Wang, S. Wu, J. Hu, Y. Jiang, J. Li, Y. Hu, Y. Han, T. Ben, B. Chen and Y. Zhang, *Chem. Sci.*, 2024, **15**, 5653–5659.

113 Y. B. He, R. Krishna and B. L. Chen, *Energy Environ. Sci.*, 2012, **5**, 9107–9120.

114 T. Ke, Q. Wang, J. Shen, J. Zhou, Z. Bao, Q. Yang and Q. Ren, *Angew. Chem., Int. Ed.*, 2020, **59**, 12725–12730.

115 L. Yang, X. Cui, Y. Zhang, Q. Yang and H. Xing, *J. Mater. Chem. A*, 2018, **6**, 24452–24458.

116 C. Yu, Z. Guo, L. Yang, J. Cui, S. Chen, Y. Bo, X. Suo, Q. Gong, S. Zhang, X. Cui, S. He and H. Xing, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218027.

117 S. Tu, D. X. Lin, J. J. Huang, L. Yu, Z. W. Liu, Z. Li and Q. B. Xia, *Microporous Mesoporous Mater.*, 2023, **354**, 112532.

118 X.-W. Zhang, H. He, Y.-W. Gan, Y. Wang, N.-Y. Huang, P.-Q. Liao, J.-P. Zhang and X.-M. Chen, *Angew. Chem., Int. Ed.*, 2024, **63**, e202317648.

119 J. Shen, X. He, T. Ke, R. Krishna and Q. Ren, *Nat. Commun.*, 2020, **11**, 6259.

120 X. Zhang, Q. C. Chen, X. F. Bai, Y. L. Zhao and J. R. Li, *Angew. Chem., Int. Ed.*, 2024, **63**, e202411744.

121 X.-W. Gu, E. Wu, J.-X. Wang, H.-M. Wen, B. Chen, B. Li and G. Qian, *Sci. Adv.*, 2023, **9**, eadh0135.

122 B. Zhu, J.-W. Cao, S. Mukherjee, T. Pham, T. Zhang, T. Wang, X. Jiang, K. A. Forrest, M. J. Zaworotko and K.-J. Chen, *J. Am. Chem. Soc.*, 2021, **143**, 1485–1492.

123 H. M. Wen, C. Y. Yu, M. Y. Liu, C. Y. Lin, B. Y. Zhao, H. Wu, W. Zhou, B. L. Chen and J. Hu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202309108.

124 X. Zhao, Y. Wang, D.-S. Li, X. Bu and P. Feng, *Adv. Mater.*, 2018, **30**, 1705189.

125 Y. Y. Xue, J. Lei, H. J. Lv, P. Liang, L. Q. Li and Q. G. Zhai, *Small*, 2024, **20**, 2311555.

126 D.-X. Xue, A. Cadiau, L. J. Weseliński, H. Jiang, P. M. Bhatt, A. Shkurenko, L. Wojtas, C. Zhijie, Y. Belmabkhout, K. Adil and M. Eddaoudi, *Chem. Commun.*, 2018, **54**, 6404–6407.

127 L. F. Yang, X. L. Cui, Q. W. Yang, S. H. Qian, H. Wu, Z. B. Bao, Z. G. Zhang, Q. L. Ren, W. Zhou, B. L. Chen and H. B. Xing, *Adv. Mater.*, 2018, **30**, 1705374–1705381.

128 Y. J. Jiang, L. Y. Wang, T. A. Yan, J. B. Hu, W. Q. Sun, R. Krishna, D. M. Wang, Z. L. Gu, D. H. Liu, X. L. Cui, H. B. Xing and Y. B. Zhang, *Chem. Sci.*, 2023, **14**, 298–309.

129 P. Zhang, L. Yang, X. Liu, J. Wang, X. Suo, L. Chen, X. Cui and H. Xing, *Nat. Commun.*, 2022, **13**, 4928.

130 C. Zhang, F. Formalik, D. Lv, F. Sha, K. O. Kirlikovali, X. Wang, X. Tang, S. Su, H. Xie, Y. Chen, Z. Li,



R. Q. Snurr and O. K. Farha, *Angew. Chem., Int. Ed.*, 2025, **64**, e202424260.

131 W. Gong, Y. Xie, A. Yamano, S. Ito, E. W. Reinheimer, J. Dong, C. D. Malliakas, D. M. Proserpio, Y. Cui and O. K. Farha, *Angew. Chem., Int. Ed.*, 2024, **63**, e202318475.

132 G. D. Wang, Y. Z. Li, R. Krishna, W. Y. Zhang, L. Hou, Y. Y. Wang and Z. Zhu, *Angew. Chem., Int. Ed.*, 2024, **63**, e202319978.

133 G.-D. Wang, Y.-Z. Li, W.-J. Shi, L. Hou, Y.-Y. Wang and Z. Zhu, *Angew. Chem., Int. Ed.*, 2022, **61**, e202205427.

134 Z.-H. Guo, X.-Q. Wu, Y.-P. Wu, D.-S. Li, G.-P. Yang and Y.-Y. Wang, *Angew. Chem., Int. Ed.*, 2025, **64**, e202421992.

135 Y.-Y. Xue, J. Lei, H.-J. Lv, P. Liang, L. Li and Q.-G. Zhai, *Small*, 2024, **20**, 2311555.

136 M.-Y. Gao, A. A. Bezrukov, B.-Q. Song, M. He, S. J. Nikkhah, S.-Q. Wang, N. Kumar, S. Darwish, D. Sensharma, C. Deng, J. Li, L. Liu, R. Krishna, M. Vandichel, S. Yang and M. J. Zaworotko, *J. Am. Chem. Soc.*, 2023, **145**, 11837–11845.

137 Y.-Z. Li, G.-D. Wang, R. Krishna, Q. Yin, D. Zhao, J. Qi, Y. Sui and L. Hou, *Chem. Eng. J.*, 2023, **466**, 143056.

138 G. D. Wang, R. Krishna, Y. Z. Li, W. J. Shi, L. Hou, Y. Y. Wang and Z. Zhu, *Angew. Chem.*, 2022, **61**, e202213015.

139 L. Li, L. Guo, F. Zheng, Z. Zhang, Q. Yang, Y. Yang, Q. Ren and Z. Bao, *ACS Appl. Mater. Interfaces*, 2020, **12**, 17147–17154.

140 Y. Jiang, L. Wang, T. Yan, J. Hu, W. Sun, R. Krishna, D. Wang, Z. Gu, D. Liu, X. Cui, H. Xing and Y. Zhang, *Chem. Sci.*, 2023, **14**, 298–309.

141 Y. Jiang, L. Wang, J. Hu, R. Krishna, B. Chen and Y. Zhang, *Adv. Mater.*, 2024, **36**, 2311140.

142 J. W. Ren, X. Dyosiba, N. M. Musyoka, H. W. Langmi, M. Mathe and S. J. Liao, *Coord. Chem. Rev.*, 2017, **352**, 187–219.

143 M. Rubio-Martinez, C. Avci-Camur, A. W. Thornton, I. Imaz, D. Maspoch and M. R. Hill, *Chem. Soc. Rev.*, 2017, **46**, 3453–3480.

144 G. R. Cai, X. Ma, M. Kassymova, K. Sun, M. L. Ding and H. L. Jiang, *ACS Cent. Sci.*, 2021, **7**, 1434–1440.

145 D. Chakraborty, A. Yurdusen, G. Mouchaham, F. Nouar and C. Serre, *Adv. Funct. Mater.*, 2024, **34**, 2309089.

146 M. Taddei, D. A. Steitz, J. A. van Bokhoven and M. Ranocchiari, *Chem.-Eur. J.*, 2016, **22**, 3245–3249.

147 T. Gao, H. J. Tang, S. Y. Zhang, J. W. Cao, Y. N. Wu, J. Chen, Y. Wang and K. J. Chen, *J. Solid State Chem.*, 2021, **303**, 122547.

148 U. Mueller, M. Schubert, F. Teich, H. Puetter, K. Schierle-Arndt and J. Pastre, *ChemInform*, 2006, **37**, 626–636.

149 J.-B. Lin, T. T. T. Nguyen, R. Vaidhyanathan, J. Burner, J. M. Taylor, H. Durekova, F. Akhtar, R. K. Mah, O. Ghaffari-Nik, S. Marx, N. Fylstra, S. S. Iremonger, K. W. Dawson, P. Sarkar, P. Hovington, A. Rajendran, T. K. Woo and G. K. H. Shimizu, *Science*, 2021, **374**, 1464–1469.

150 S. Geng, E. Lin, X. Li, W. Liu, T. Wang, Z. Wang, D. Sensharma, S. Darwish, Y. H. Andalousi, T. Pham, P. Cheng, M. J. Zaworotko, Y. Chen and Z. Zhang, *J. Am. Chem. Soc.*, 2021, **143**, 8654–8660.

151 B. Zhou, Q. Li, Q. Zhang, J. Duan and W. Jin, *J. Membr. Sci.*, 2020, **597**, 117772.

152 D. Crawford, J. Casaban, R. Haydon, N. Giri, T. McNally and S. L. James, *Chem. Sci.*, 2015, **6**, 1645–1649.

153 C. Xu, S. De, A. M. Balu, M. Ojeda and R. Luque, *Chem. Commun.*, 2015, **51**, 6698–6713.

154 M. Vadivelu, S. Sugirdha, P. Dheen Kumar, Y. Arun and K. Karthikeyan, *Green Chem.*, 2017, **19**, 3601–3610.

155 Y. Wang, H. Wang, Y. Jiang, C. Zhang, J. Shao and D. Xu, *Green Chem.*, 2017, **19**, 1674–1677.

156 Q. Cao, R. T. Fallis, I. A. Browne and L. Duncan, *ChemSusChem*, 2019, **12**, 2554–2557.

157 A. Pichon, A. Lazuen-Garay and S. L. James, *CrystEngComm*, 2006, **8**, 211–214.

158 Y. Liang, G. Xie, K.-K. Liu, M. Jin, Y. Chen, X. Yang, Z.-J. Guan, H. Xing and Y. Fang, *Angew. Chem., Int. Ed.*, 2025, **64**, e202416884.

159 T. D. Bennett, S. Cao, J. C. Tan, D. A. Keen, E. G. Bithell, P. J. Beldon, T. Friscic and A. K. Cheetham, *J. Am. Chem. Soc.*, 2011, **133**, 14546–14549.

160 O. M. Yaghi, R. Jernigan, H. Li, C. E. Davis and T. L. Groy, *J. Chem. Soc., Dalton Trans.*, 1997, 2383–2384.

161 S.-N. Kim, Y.-R. Lee, S.-H. Hong, M.-S. Jang and W.-S. Ahn, *Catal. Today*, 2015, **245**, 54–60.

162 T. K. Vo, V. N. Le, K. S. Yoo, M. Song, D. Kim and J. Kim, *Cryst. Growth Des.*, 2019, **19**, 4949–4956.

163 K. Pobłocki, J. Drzeżdżon, B. Gawdzik and D. Jacewicz, *Green Chem.*, 2022, **24**, 9402–9427.

164 U. Mueller, M. Schubert, F. Teich, H. Puetter, K. Schierle-Arndt and J. Pastre, *J. Mater. Chem.*, 2006, **16**, 626–636.

165 J. Britton and C. L. Raston, *Chem. Soc. Rev.*, 2017, **46**, 1250.

166 M. Guidi, P. H. Seeberger and K. Gilmore, *Chem. Soc. Rev.*, 2020, **49**, 8910.

167 C. McKinstry, R. J. Cathcart, E. J. Cussen, A. J. Fletcher, S. V. Patwardhan and J. Sefcik, *Chem. Eng. J.*, 2016, **285**, 718–725.

168 G. H. Albuquerque, R. C. Fitzmorris, M. Ahmadi, N. Wannenmacher, P. K. Thallapally, B. P. McGrail and G. S. Herman, *CrystEngComm*, 2015, **17**, 5502–5510.

169 H. Ren and T. X. Wei, *Chemelectrochem*, 2022, **9**, e202200196.

170 N. Campagnol, T. R. C. Van Assche, M. Y. Li, L. Stappers, M. Dinca, J. F. M. Denayer, K. Binnemans, D. E. De Vos and J. Fransaer, *J. Mater. Chem. A*, 2016, **4**, 3914–3925.

