

Cite this: *Chem. Sci.*, 2025, 16, 18135

All publication charges for this article have been paid for by the Royal Society of Chemistry

## Coexisting gases regulate the rates of water adsorption by a flexible one-dimensional coordination polymer

AnQi Wang,<sup>a</sup> Xin Zheng,<sup>ab</sup> Yuki Saito,<sup>ab</sup> Arata Tateishi,<sup>a</sup> Yuan Huang,<sup>b</sup> Yuichi Kamiya,<sup>ib</sup> Hiroyasu Sato,<sup>c</sup> Atsushi Kondo,<sup>ib</sup> Kiyonori Takahashi,<sup>ib aeg</sup> Takayoshi Nakamura<sup>aef</sup> and Shin-ichiro Noro<sup>ib \*ab</sup>

A novel flexible one-dimensional coordination polymer  $[\text{Cu}_2(3\text{-OH-bza})_3(\text{AcO})(\text{pyr})]$  (**1**, 3-OH-bza = 3-hydroxybenzoate, AcO = acetate, pyr = pyrimidine) was found to adsorb water at rates that are influenced by the nature of coexisting gases. Upon exposure to a flow of water vapor containing gas, **1** displays a  $\text{H}_2\text{O}$  adsorption rate that is decelerated to a greater extent by  $\text{CO}_2$  than by  $\text{N}_2$ . Key to this phenomenon is the observation that **1** undergoes a structural change upon reversible and selective adsorption of  $\text{H}_2\text{O}$ . This finding serves as the basis of a new strategy for designing porous materials for highly efficient separation and storage applications.

Received 4th March 2025  
Accepted 26th August 2025

DOI: 10.1039/d5sc01699a

rsc.li/chemical-science

## Introduction

Porous materials exhibit interesting properties that enable their practical use in storage, mixture separation, catalysis, and ion exchange applications. Recently, metal-organic frameworks (MOFs) and porous coordination polymers (PCPs) have attracted attention as next-generation porous materials because of their high porosity, high structural designability, versatility and flexibility.<sup>1–3</sup> A variety of properties of MOFs/PCPs have been thoroughly investigated, the results of which have demonstrated that their storage and mixture separation capabilities excel those of other porous materials.<sup>4–10</sup>

Several key parameters need to be considered when designing excellent storage and separation materials based on MOFs/PCPs, with adsorption rate being among the most important in the context of practical applications. Break-through experiments are often used to evaluate the adsorption and separation performances of materials under gas flow conditions. In these assessments, MOFs/PCPs need to be shaped because their use in fine particle form causes a large pressure drop. However, shaping MOFs/PCPs often significantly

affects adsorption rates.<sup>11,12</sup> On the other hand, rapid adsorption by these materials causes a decrease in cycle times, resulting in an increased throughput.<sup>13</sup> In this regard, materials that operate using kinetic (or diffusion rate) separation are more efficient than those that rely on equilibrium separation.

Several studies have been conducted to develop approaches that increase and control the adsorption rates of MOFs/PCPs. For example, Vogel, Watanabe, *et al.* described a structurally hierarchical MOF that displays rapid gas adsorption. Specifically, these workers observed that a pellet packed with supraparticles of the zeolitic imidazolate framework-8 (ZIF-8) undergoes  $\text{N}_2$  adsorption at a rate that is 30 times faster than that of an unstructured ZIF-8 powder pellet.<sup>11</sup> In another effort, Long *et al.* systematically investigated the  $\text{CO}_2$  adsorption kinetics of the three-dimensional (3D) MOF,  $[\text{Mg}_2(\text{dobpdc})]$  (dobpdc = 4,4'-dioxidobiphenyl-3,3'-dicarboxylate) and its diamine-appended derivatives. The findings show that the  $\text{CO}_2$  adsorption rates are enhanced and induction periods are decreased as the temperature decreases.<sup>14</sup> Moreover, we have recently demonstrated that coating surfaces of MOF particles with a glassy nonporous coordination polymer (g-NCP) can be employed to alter adsorption rates. Specifically, we observed that the 3D MOF,  $[\text{Cu}_2(\text{pzdc})_2(\text{pyz})]$  (CPL-1, pzdc = 2,3-pyrazinedicarboxylate and pyz = pyrazine),<sup>15,16</sup> has similar rates for adsorption of  $\text{CO}_2$ ,  $\text{N}_2\text{O}$  and  $\text{C}_2\text{H}_4$ , whereas composites composed of this MOF and g-NCP,  $[\text{Cu}(\text{bib})_{2.5}] \cdot 2\text{NTf}_2$  (bib = 1,4-bisimidazole butane and  $\text{NTf}_2$  = bis(trifluoromethylsulfonyl)amide), display clearly different rates for adsorption of these gases. Differences in these materials are a consequence of the occurrence of gas diffusion within the g-NCP of the composite  $[\text{Cu}_2(\text{pzdc})_2(\text{pyz})]/\text{g-NCP}$  through a solution-diffusion mechanism.<sup>17</sup>

Most previous studies aimed at increasing and controlling adsorption rates, including those described above, have focused

<sup>a</sup>Graduate School of Environmental Science, Hokkaido University, Sapporo 060-0810, Japan. E-mail: noro@ees.hokudai.ac.jp

<sup>b</sup>Faculty of Environmental Earth Science, Hokkaido University, Sapporo 060-0810, Japan

<sup>c</sup>Rigaku Corporation, Akishima 196-8666, Japan

<sup>d</sup>Faculty of Science and Technology, Oita University, Oita 870-1192, Japan

<sup>e</sup>Research Institute for Electronic Science, Hokkaido University, Sapporo 001-0020, Japan

<sup>f</sup>Department of Chemistry, Graduate School of Advanced Science and Engineering, Hiroshima University, 1-3-1, Kagamiyama, Higashi-hiroshima 739-8526, Japan

<sup>g</sup>Faculty of Advanced Science and Technology, Kumamoto University, Kumamoto, 860-8555, Japan

on changing the micro- or macrostructures of MOFs. In the current investigation described below, we developed an MOF in which the H<sub>2</sub>O adsorption rate is governed by the nature of coexisting gases. Specifically, the novel flexible one-dimensional (1D) coordination polymer [Cu<sub>2</sub>(3-OH-bza)<sub>3</sub>(AcO)(pyr)] (**1**, 3-OH-bza = 3-hydroxybenzoate, AcO = acetate, pyr = pyrimidine) adsorbs H<sub>2</sub>O when exposed to not only pure H<sub>2</sub>O vapor but also when exposed to flows of carrier gas (N<sub>2</sub> and CO<sub>2</sub>) containing water vapor. Also, water adsorption promotes a structural change of **1**. Measurements have shown that the rate of H<sub>2</sub>O adsorption under flow conditions is higher when N<sub>2</sub> rather than CO<sub>2</sub> is the carrier gas because of a difference in the diffusion coefficient of the binary gas system.

## Results and discussion

[Cu<sub>2</sub>(3-OH-bza)<sub>3</sub>(AcO)(pyr)]·3H<sub>2</sub>O (**1**·3H<sub>2</sub>O) was prepared by the reaction of Cu(AcO)<sub>2</sub>·H<sub>2</sub>O, 3-hydroxybenzoic acid (3-OH-Hbza), and pyr in a solution of H<sub>2</sub>O and MeOH. X-ray crystallographic

analysis (Fig. 1 and Table S1) shows that **1**·3H<sub>2</sub>O has an unprecedented 1D chain structure composed of paddlewheel dimers. In contrast to most 1D coordination polymers that are created from one type of paddlewheel dimer,<sup>18–20</sup> **1**·3H<sub>2</sub>O contains an alternating array of two types of paddlewheel dimers, [Cu<sub>2</sub>(3-OH-bza)<sub>4</sub>] and [Cu<sub>2</sub>(3-OH-bza)<sub>2</sub>(AcO)<sub>2</sub>] (Fig. 1(a) and (b)). Furthermore, [Cu<sub>2</sub>(3-OH-bza)<sub>2</sub>(AcO)<sub>2</sub>] is a heteroleptic dimer containing mixed carboxylate groups. It is interesting to note that examples of 1D coordination polymers composed of heteroleptic dimers are limited, although a new method to synthesize heteroleptic paddlewheel dimers has been recently described by Miyasaka *et al.*<sup>21</sup> The two types of dimers in **1**·3H<sub>2</sub>O are alternately bridged by pyr to form a 1D zigzag chain oriented along the *c* axis (Fig. 1(b)). The 1D chains aggregate through COO⋯HO hydrogen bonding interactions (O⋯O distance = 2.750(4) Å, Fig. 1(c)) to form a porous crystal with one-dimensional channels aligned along the *a*-axis (Fig. 1(d) and S1). The amount of void space in the structure was calculated using Mercury software to be 11%. Guest H<sub>2</sub>O molecules are present in these channels positioned by rich hydrogen bonding interactions not only between the H<sub>2</sub>O molecules but also between the H<sub>2</sub>O and COO/OH groups of the carboxylate ligands (Fig. 1(e)).

The thermal properties of **1**·3H<sub>2</sub>O were evaluated using a thermogravimetric-differential thermal analysis (TG-DTA) measurement. As shown in Fig. 2(a), guest H<sub>2</sub>O molecules in the material are released in the range of room temperature to *ca.* 50 °C. The weight loss at 50 °C is 5.4%, corresponding to 2.2 mol of H<sub>2</sub>O per 1 mol of **1**. The weight loss, which is slightly less than 3 mol per 1 mol calculated from analysis of the crystal structure of **1**·3H<sub>2</sub>O, is caused by gradual desorption of water in air. Desolvated **1** formed by H<sub>2</sub>O release is stable up to *ca.* 220 °C.

The attenuated total reflection (ATR)-infrared (IR) spectrum of desolvated **1** was found to contain only an O–H vibration band associated with 3-OH-bza ligands, supporting the conclusion that all guest H<sub>2</sub>O molecules are released during thermally promoted dehydration (Fig. S4(a)).

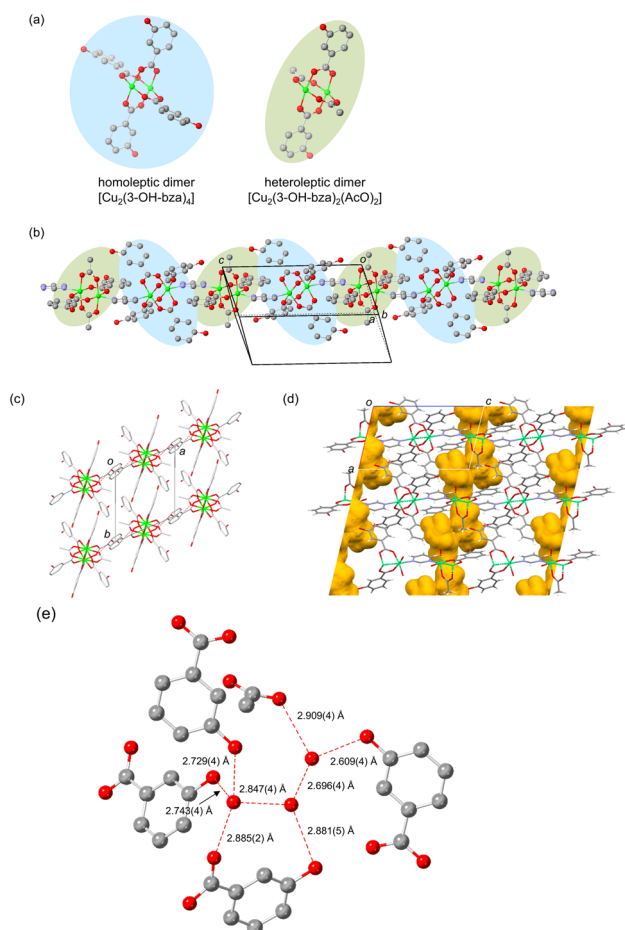


Fig. 1 Crystal structure of **1**·3H<sub>2</sub>O. (a) Two types of contributing paddlewheel dimers forming (b) a 1D zigzag chain structure. (c) Packing of the chains, (d) one-dimensional channel structure, and (e) hydrogen bonds involving guest water molecules. The green, blue, gray, and red represent copper, nitrogen, carbon, and oxygen, respectively. Hydrogen atoms are omitted for clarity except in (d). Red dashed lines represent hydrogen bonds.

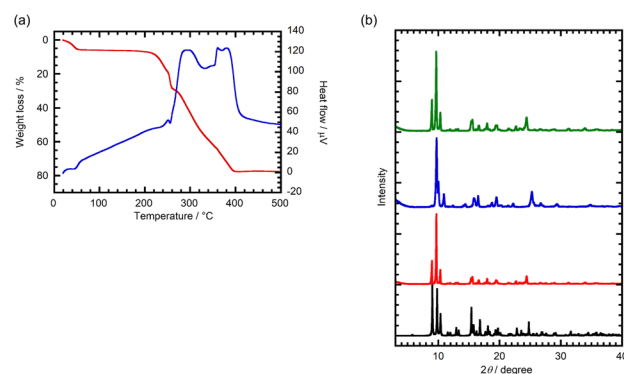


Fig. 2 (a) TG (red)-DTA (blue) curves of **1**·3H<sub>2</sub>O. (b) Simulated PXRD pattern of **1**·3H<sub>2</sub>O (black), observed PXRD patterns of as-synthesized **1**·3H<sub>2</sub>O (red), **1** obtained after heating of **1**·3H<sub>2</sub>O at 100 °C under vacuum (blue), and **1**·3H<sub>2</sub>O obtained after H<sub>2</sub>O exposure to dehydrated **1** (green).



Powder X-ray diffraction (PXRD) analysis was employed to evaluate the structure of **1** after H<sub>2</sub>O removal. Desolvated **1** displays a PXRD pattern that differs from both the simulated and real patterns of **1**·3H<sub>2</sub>O (Fig. 2(b)), suggesting that crystal-to-crystal structural transformation takes place during the release of H<sub>2</sub>O. Finally, exposure of desolvated **1** to H<sub>2</sub>O causes regeneration of the hydrated form **1**·3H<sub>2</sub>O (Fig. 2(b)), revealing that H<sub>2</sub>O adsorption/desorption occurs reversibly in conjunction with a structural change.

We succeeded in carrying out structural characterization of desolvated **1** using electron diffraction analysis (EDA). Data obtained from EDA experiments show that after the release of H<sub>2</sub>O molecules, **1** retains its 1D zigzag chain structure (Fig. S2). However, the pattern of interchain hydrogen-bonding interactions is significantly changed upon dehydration. Specifically, in **1**·3H<sub>2</sub>O, the interchain hydrogen bonds are oriented in the *ac* plane (Fig. 1(c)), while those in dehydrated **1** are aligned in a 3D manner presumably for crystal structural stabilization (Fig. 3). Thus, reorganization of the hydrogen bonding network in **1**·3H<sub>2</sub>O takes place during dehydration to promote more dense packing of the chains creating a very small void space of 2% (Fig. S3). Also, the purity of the sample of **1** produced by thermal treatment of **1**·3H<sub>2</sub>O was confirmed using PXRD. The cell parameters arising from Le Bail fitting for the PXRD pattern of the desolvated **1** agree well with those obtained employing EDA (Fig. S7), indicating that the sample includes only a single structure shown in Fig. 3.

Experiments were conducted to assess the detailed H<sub>2</sub>O adsorption properties of desolvated **1**. In Fig. 4(a) are given the H<sub>2</sub>O, MeOH and EtOH adsorption/desorption isotherms of this material. The results indicate that a gradual increase in the amount of H<sub>2</sub>O adsorbed occurs in the  $P/P_0$  range from 0 to 0.41, and that the total amount of H<sub>2</sub>O adsorbed reaches 0.57 mol mol<sup>−1</sup>. Above  $P/P_0 = 0.41$ , a sudden increase in H<sub>2</sub>O adsorption takes place reaching a saturated state of *ca.* 3 mol mol<sup>−1</sup>, which is consistent with formation of **1**·3H<sub>2</sub>O. In the desorption process, **1**·3H<sub>2</sub>O undergoes a rapid decrease in the amount of adsorbed H<sub>2</sub>O below  $P/P_0 = 0.34$ , and the adsorption and

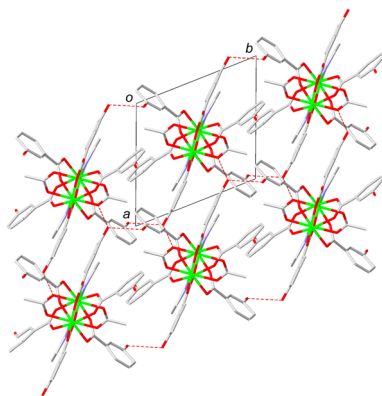


Fig. 3 EDA-derived packing structure of chains in dehydrated **1**. The green, blue, grey, and red colors represent copper, nitrogen, carbon, and oxygen, respectively. Hydrogen atoms are omitted for clarity. Red dashed lines represent hydrogen bonds.

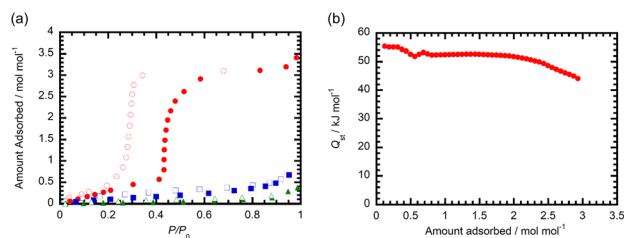


Fig. 4 (a) H<sub>2</sub>O (red circles, 298 K), MeOH (blue squares, 288 K), and EtOH (green triangle, 298 K) adsorption (filled symbols)/desorption (open symbols) isotherms of **1**. (b) Isothermic heat of water adsorption for **1**.

desorption isotherms have a large hysteresis. These observations are consistent with the PXRD results indicating that **1** undergoes reversible H<sub>2</sub>O adsorption/desorption along with an accompanying structural change that can be described as H<sub>2</sub>O gated sorption. The results of experiments aimed at elucidating adsorption selectivity suggest that **1** adsorbs small amounts of not only N<sub>2</sub> (77 K) and CO<sub>2</sub> (195 and 298 K) gas (Fig. S8 and S9) but also MeOH and EtOH vapor (Fig. 3(a)). Overall, these observations indicate that **1** has high selectivity for H<sub>2</sub>O over other small molecules.

We next estimated the isosteric heat of adsorption ( $Q_{st}$ ) of **1** for H<sub>2</sub>O using isotherms obtained at various temperatures (Fig. S10). The  $Q_{st}$  values for **1** before and after rapid increases in adsorption amounts were found to be 52–55 and 44–53 kJ mol<sup>−1</sup> (Fig. 3(b)), which are larger than the latent energy of water (40.7 kJ mol<sup>−1</sup>). The  $Q_{st}$  value of 53 kJ mol<sup>−1</sup> determined after a rapid increase in the amount of adsorbed H<sub>2</sub>O is slightly higher than that found earlier for [Zr<sub>6</sub>O<sub>4</sub>(OH)<sub>4</sub>(MTB)<sub>2</sub>(HCOO)<sub>4</sub>(H<sub>2</sub>O)<sub>2</sub>] (MOF-841, MTB = 4,4',4'',4'''-methanetetrayltetrabenzoate) (50 kJ mol<sup>−1</sup>).<sup>22</sup>

The H<sub>2</sub>O adsorption/desorption behaviour of **1** was also evaluated using TG-DTA under humidified and dry N<sub>2</sub> conditions. As the plot in Fig. 5 shows, H<sub>2</sub>O adsorption at 300 K reaches a plateau after 40 min, at which the amount of H<sub>2</sub>O adsorbed is 2.86 mol mol<sup>−1</sup>. This value is very close to the *ca.* 3 mol mol<sup>−1</sup> value arising from the H<sub>2</sub>O adsorption isotherm. These results suggest that **1** adsorbs H<sub>2</sub>O almost exclusively even under mixed N<sub>2</sub>–H<sub>2</sub>O flow conditions. After switching from wet to dry N<sub>2</sub> gas, the amount of adsorbed H<sub>2</sub>O decreases to zero

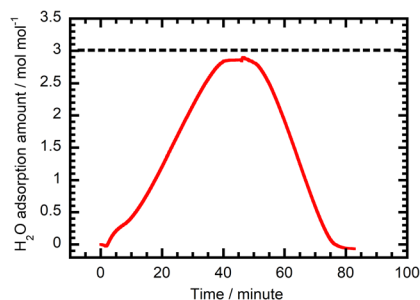


Fig. 5 Time dependence of the amount of H<sub>2</sub>O adsorbed by **1** under humidified N<sub>2</sub> (0–46 min) and dry N<sub>2</sub> (46–83 min) at 300 K.



at 80 min. When the carrier gas is CO<sub>2</sub>, similar H<sub>2</sub>O content changes take place (humidified CO<sub>2</sub> flow reaching 2.86 mol mol<sup>-1</sup> and dry flow back to zero, Fig. S12), suggesting that the nature of the carrier gas has no effect on the amount of H<sub>2</sub>O adsorbed.

We also determined the rates of H<sub>2</sub>O adsorption by **1** using IR spectroscopy rather than TG-DTA because of the difficulties in controlling temperatures near room temperature and performing rate measurements inside the TG-DTA chamber. Comparison of ATR-IR spectra of **1**·3H<sub>2</sub>O and **1** indicates that dramatic differences exist in the 1150–1300 cm<sup>-1</sup> range (Fig. S4(b)), which are associated with C–O (hydroxyl oxygen) vibration bands of 3-OH-bza ligands (Fig. S5). These observations are consistent with the results of the crystal structure analysis that suggest that dehydration alters the hydrogen bonding patterns of the OH groups. Therefore, this wavelength range was employed to monitor the rates of H<sub>2</sub>O adsorption by **1**. Inspection of the spectra in Fig. 6(a) shows that dramatic changes in peak intensities occur when **1** is subjected to a humid N<sub>2</sub> gas flow because of structural changes promoted by H<sub>2</sub>O adsorption. The time course for changes in the intensity of the peak at 1227 cm<sup>-1</sup> given in Fig. 6(b) indicates that the adsorption process reaches saturation after 15 min.

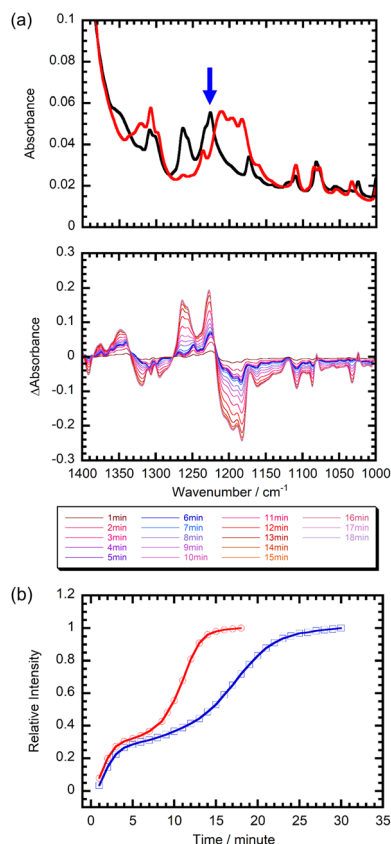


Fig. 6 (a) ATR-IR spectra of **1** (red) and **1**·3H<sub>2</sub>O (black) and difference IR spectra of **1** under wet N<sub>2</sub> at 303 K over 1 min intervals with the humid N<sub>2</sub> gas flow beginning at 0 min. (b) Time dependence of relative intensity for a peak at 1227 cm<sup>-1</sup> (blue arrow in (a) indicates the wavenumber monitored) under humid N<sub>2</sub> (red) and humid CO<sub>2</sub> (blue) at 303 K in **1**.

Surprisingly, the rate of H<sub>2</sub>O adsorption by **1** is carrier gas dependent. Specifically, although the spectral changes promoted by subjecting **1** to a flow of humidified CO<sub>2</sub> are the same as those brought about by humid N<sub>2</sub> (Fig. S13), they occur more slowly than in the former experiment (especially, after 7 min) to reach the saturation point after 25 min (Fig. 6(b)). The adsorption rates ( $k_{\text{N}_2\text{-ad}}$  and  $k_{\text{CO}_2\text{-ad}}$ ) calculated using the time dependency data were found to be 0.11 and 0.055 min<sup>-1</sup> for wet N<sub>2</sub> and CO<sub>2</sub> gases, respectively (Fig. S14). Notably, H<sub>2</sub>O desorption under dry N<sub>2</sub> and CO<sub>2</sub> flows occurs at similar rates ( $k_{\text{N}_2\text{-de}} = 0.80$  and  $k_{\text{CO}_2\text{-de}} = 0.85$  min<sup>-1</sup>) and more rapidly than adsorption of H<sub>2</sub>O (Fig. S15–S18).

To confirm the generality of this phenomenon, similar water adsorption experiments were performed on another flexible coordination polymer. The two-dimensional (2D) [Cu(CF<sub>3</sub>SO<sub>3</sub>)<sub>2</sub>(bpp)<sub>2</sub>] (bpp = 1,3-bis(4-pyridyl)propane) shows reversible H<sub>2</sub>O adsorption/desorption with temporary expansion of the 2D layer.<sup>23</sup> This coordination polymer adsorbs *ca.* 1 mol mol<sup>-1</sup> of H<sub>2</sub>O at 298 K (ref. 23) but adsorbs only a small amount of CO<sub>2</sub> at the same temperature (Fig. S11). The ATR-IR spectrum of [Cu(CF<sub>3</sub>SO<sub>3</sub>)<sub>2</sub>(bpp)<sub>2</sub>] contains vibration bands for CF<sub>3</sub>SO<sub>3</sub> anions in the 1000–1300 cm<sup>-1</sup> range (Fig. S19).<sup>23</sup> Analysis of the time-dependence of the relative intensity of the IR peak at 1163 cm<sup>-1</sup> (Fig. S19–S21) indicates that the rate of H<sub>2</sub>O adsorption under a humid CO<sub>2</sub> gas flow is slower than that under a humid N<sub>2</sub> gas flow ( $k_{\text{N}_2\text{-ad}}$  and  $k_{\text{CO}_2\text{-ad}} = 0.39$  and 0.29 min<sup>-1</sup> for wet N<sub>2</sub> and CO<sub>2</sub> gases, respectively, Fig. S22). However, the difference in the adsorption rate, calculated using the time dependency data ( $k_{\text{N}_2\text{-ad}}/k_{\text{CO}_2\text{-ad}} = 1.34$ ), is smaller than that observed for **1** ( $k_{\text{N}_2\text{-ad}}/k_{\text{CO}_2\text{-ad}} = 2.00$ ), indicating that the degree of a coexisting guest effect depends on nature of the structure. On the other hand, the desorption rates are almost the same ( $k_{\text{N}_2\text{-de}}$  and  $k_{\text{CO}_2\text{-de}} = 0.42$  and 0.41 min<sup>-1</sup>) regardless of the type of coexisting gas (Fig. S23–S26).

Several possible sources for the carrier gas dependence of the rate of H<sub>2</sub>O adsorption were considered. First, we realized that CO<sub>2</sub> has high solubility in H<sub>2</sub>O, and that dissolution results in the formation of carbonic acid (H<sub>2</sub>CO<sub>3</sub>) and its conjugate bases (HCO<sub>3</sub><sup>-</sup> and CO<sub>3</sub><sup>2-</sup>). Moreover, Kitagawa, Onoe, *et al.* reported that CO<sub>3</sub><sup>2-</sup> anions form in the specific sub-nm space of a 1D uneven-structured C<sub>60</sub> polymer film subjected to CO<sub>2</sub> and H<sub>2</sub>O at room temperature.<sup>24</sup> While H<sub>2</sub>CO<sub>3</sub> and CO<sub>3</sub><sup>2-</sup> display bands at *ca.* 1180 and 1363 cm<sup>-1</sup>, respectively,<sup>24</sup> these researchers observed a band at only 1370 cm<sup>-1</sup> in the IR spectrum of the C<sub>60</sub> polymer film after exposure to atmospheric air (*ca.* 14% humidity), suggesting the presence of CO<sub>3</sub><sup>2-</sup> anions.

It is possible that H<sub>2</sub>CO<sub>3</sub>, HCO<sub>3</sub><sup>-</sup> and CO<sub>3</sub><sup>2-</sup>, formed transiently during H<sub>2</sub>O adsorption under a humid CO<sub>2</sub> gas flow, decrease the H<sub>2</sub>O adsorption rate owing to a blocking effect caused by strong H<sub>2</sub>CO<sub>3</sub>/HCO<sub>3</sub><sup>-</sup>/CO<sub>3</sub><sup>2-</sup>-interactions with framework sites in **1** (Fig. 7(a(i))). To determine if the formation of H<sub>2</sub>CO<sub>3</sub>, HCO<sub>3</sub><sup>-</sup> and CO<sub>3</sub><sup>2-</sup> occurs under a humid CO<sub>2</sub> flow, the 1000–1400 cm<sup>-1</sup> region in IR spectra of **1** under humid CO<sub>2</sub> and humid N<sub>2</sub> flows was analyzed. However, no differences were found to exist in spectra obtained using both flows (Fig. 6(a) and S13).





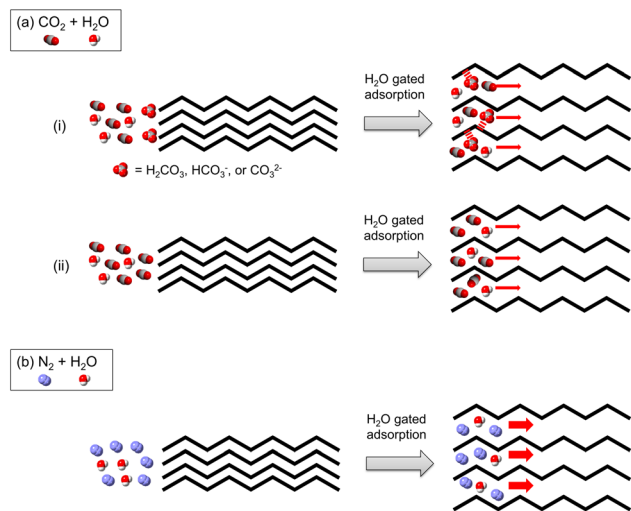


Fig. 7 Possible source of the coexisting gas dependency of the H<sub>2</sub>O adsorption rate. (a) Temporary formation of H<sub>2</sub>CO<sub>3</sub>/HCO<sub>3</sub><sup>-</sup>/CO<sub>3</sub><sup>2-</sup> (i) and temporary occupancy of CO<sub>2</sub> and H<sub>2</sub>O in the pores (ii) under humid CO<sub>2</sub> flow conditions. (b) Temporary occupancy of N<sub>2</sub> and H<sub>2</sub>O in the pores under wet N<sub>2</sub> flow conditions.

Next, we focused on diffusion coefficients for binary gas systems because it is known that this parameter varies with the type of coexisting gas in a mixture. For example, the diffusion coefficient of the CO<sub>2</sub>–H<sub>2</sub>O binary (equimolar mixture) system at 293.15 K is 0.162 cm<sup>2</sup> s<sup>-1</sup>, a value that is considerably lower than that of the N<sub>2</sub>–H<sub>2</sub>O binary system (0.242 cm<sup>2</sup> s<sup>-1</sup> at 293.15 K).<sup>25</sup> Therefore, the greater propensity of CO<sub>2</sub> to decelerate water diffusion is the likely cause of the lower rate of H<sub>2</sub>O adsorption by **1** under a CO<sub>2</sub>–H<sub>2</sub>O flow compared to a N<sub>2</sub>–H<sub>2</sub>O flow (Fig. 7(a(ii)) and (b)).

On the other hand, a carrier gas independence of the H<sub>2</sub>O desorption rates was observed, suggesting that the H<sub>2</sub>O adsorption and desorption processes in the presence of coexisting guests are considerably different. During H<sub>2</sub>O desorption, CO<sub>2</sub> and N<sub>2</sub> do not diffuse through pores as shown in Fig. 8, which is the likely cause of the observed similar desorption rates under N<sub>2</sub> and CO<sub>2</sub> flows.

If correctly interpreted, this finding would be the first example of a coexisting guest effect on rates of adsorption of

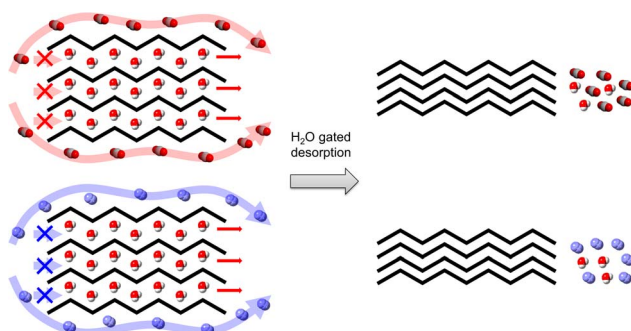


Fig. 8 Scheme of the possible source of the coexisting gas independence of the H<sub>2</sub>O desorption rate.

guests by materials. This proposal raises the question whether CO<sub>2</sub> and/or N<sub>2</sub> gases diffuse through pores of **1** (Fig. 7) along with H<sub>2</sub>O while not being adsorbed. Previously, we investigated the influence of co-adsorbates on CO<sub>2</sub> gated sorption in flexible MOFs.<sup>26</sup> We observed that [Zn<sub>2</sub>(DiP-bdc)<sub>2</sub>(dabco)] (DiP-bdc = 2,5-diisopropoxy-1,4-benzenedicarboxylate; dabco = 1,4-diazabicyclo[2.2.2]octane) displays gated sorption of CO<sub>2</sub>, C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> in conjunction with a structural change from a narrow pore (np) to a large pore (lp) form. In contrast, N<sub>2</sub> and CH<sub>4</sub> do not promote gate opening, which results in low adsorption of these gases. However, results of co-adsorption measurements demonstrated that after CO<sub>2</sub> induced gate opening, CH<sub>4</sub> co-adsorbs in the lp phase. In other words, CH<sub>4</sub> enters pores of this MOF along with CO<sub>2</sub>. Based on these previous results, it is reasonable to propose that unlike H<sub>2</sub>O, CO<sub>2</sub> and N<sub>2</sub> occupy the pores only transiently. However, further investigations are needed to confirm that transient occupation in the pores occurs.

## Conclusions

In conclusion, we investigated features of H<sub>2</sub>O adsorption by the 1D coordination polymer [Cu<sub>2</sub>(3-OH-bza)<sub>3</sub>(AcO)(pyr)] (**1**) under gas flow conditions. The results of single-crystal X-ray analysis, EDA, and gas and vapor adsorption/desorption measurements show that the coordination polymer displays reversible and selective H<sub>2</sub>O adsorption in conjunction with a structural change. Kinetic analysis carried out by using *in situ* IR spectroscopy indicates that the rate of H<sub>2</sub>O adsorption under a humid CO<sub>2</sub> gas flow is slower than that under a humid N<sub>2</sub> gas flow. This trend is consistent with the diffusion coefficients of the respective binary gas systems. Specifically, the considerably lower diffusion coefficient of the CO<sub>2</sub>–H<sub>2</sub>O mixture compared to that of the corresponding N<sub>2</sub>–H<sub>2</sub>O mixture is responsible for the greater propensity of CO<sub>2</sub> to decelerate water adsorption by **1**. In addition, this coexisting guest effect was observed in another flexible 2D coordination polymer. While it is obvious that the diffusion coefficient of the binary gas system depends on the types of gases, it is noteworthy that the phenomenon was observed for gas flows within a very narrow (<1 nm) space. Therefore, this finding suggests that a new strategy exists to control gas/vapor adsorption rates of porous materials, and that the approach might be applicable to the development of efficient gas storage and separation materials.

## Author contributions

S. N. conceptualized the project. A. W., X. Z., Y. S., A. T., Y. H., K. T. and T. N. contributed to data collection and formal analyses. S. N., Y. K. and Y. H. set up and performed *in situ* IR spectra measurements. H. S. performed electron diffraction measurements and analyses. A. K. performed Le Bail fitting of the PXRD pattern. S. N. wrote the manuscript, and all the authors approved the final version.



## Conflicts of interest

There are no conflicts to declare.

## Data availability

CCDC 2428329 (1·3H<sub>2</sub>O) and 2428330 (1) contain the supplementary crystallographic data for this paper.<sup>27a,b</sup>

Supplementary information is available. See DOI: <https://doi.org/10.1039/d5sc01699a>.

## Notes and references

- 1 S. Kitagawa, R. Kitaura and S. Noro, *Angew. Chem., Int. Ed.*, 2004, **43**, 2334.
- 2 H. Furukawa, K. E. Cordova, M. O'Keeffe and O. M. Yaghi, *Science*, 2013, **341**, 1230444.
- 3 L. Carlucci, G. Ciani, D. M. Proserpio, T. G. Mitina and V. A. Blatov, *Chem. Rev.*, 2014, **114**, 7557.
- 4 Y. Peng, V. Krungleviciute, I. Eryazici, J. T. Hupp, O. K. Farha and T. Yildirim, *J. Am. Chem. Soc.*, 2013, **135**, 11887.
- 5 K. Suresh, D. Aulakh, J. Purewal, D. J. Siegel, M. Veenstra and A. J. Matzger, *J. Am. Chem. Soc.*, 2021, **143**, 10727.
- 6 W. Lu, D. D. A. Jayasinghe, M. Schröder and S. Yang, *Acc. Mater. Res.*, 2024, **5**, 1279.
- 7 T. T. T. Nguyen, J.-B. Lin, G. K. H. Shimizu and A. Rajendran, *Chem. Eng. J.*, 2022, **442**, 136263.
- 8 Y. Su, K. Otake, J.-J. Zhang, S. Horike, S. Kitagawa and C. Gu, *Nature*, 2022, **611**, 289.
- 9 P. Vervoorts, A. Schneemann, I. Hante, J. Pirillo, Y. Hijikata, T. Toyao, K. Kon, K. Shimizu, T. Nakamura, S. Noro and R. A. Fischer, *ACS Appl. Mater. Interfaces*, 2020, **12**, 9448.
- 10 Y. Tulchinsky, C. H. Hendon, K. A. Lomachenko, E. Borfecchia, B. C. Melot, M. R. Hudson, J. D. Tarver, M. D. Korzyński, A. W. Stubbs, J. J. Kagan, C. Lamberti, C. M. Brown and M. Dincă, *J. Am. Chem. Soc.*, 2017, **139**, 5992.
- 11 A. Fujiwara, J. Wang, S. Hiraide, A. Götz, M. T. Miyahara, M. Hartmann, B. A. Zubiri, E. Spiecker, N. Vogel and S. Watanabe, *Adv. Mater.*, 2023, **35**, 2305980.
- 12 Y. Chen, X. Huang, S. Zhang, S. Li, S. Cao, X. Pei, J. Zhou, X. Feng and B. Wang, *J. Am. Chem. Soc.*, 2016, **138**, 10810.
- 13 R. T. Yang, *Gas Separation by Adsorption Processes*, Butterworth-Heinemann, Boston, 1987, ch. 5, pp. 141–200.
- 14 J. D. Martell, P. J. Milner, R. L. Siegelman and J. R. Long, *Chem. Sci.*, 2020, **11**, 6457.
- 15 M. Kondo, T. Okubo, A. Asami, S. Noro, T. Yoshitomi, S. Kitagawa, T. Ishii, H. Matsuzaka and K. Seki, *Angew. Chem., Int. Ed.*, 1999, **38**, 140.
- 16 R. Kitaura, R. Matsuda, Y. Kubota, S. Kitagawa, M. Takata, T. C. Kobayashi and M. Suzuki, *J. Phys. Chem. B*, 2005, **109**, 23378.
- 17 X. Zheng, M. Kato, Y. Uemura, D. Matsumura, I. Yagi, K. Takahashi, S. Noro and T. Nakamura, *Inorg. Chem.*, 2023, **62**, 1257.
- 18 S. Takamizawa, E. Nakata, T. Akatsuka, R. Miyake, Y. Kakizaki, H. Takeuchi, G. Maruta and S. Takeda, *J. Am. Chem. Soc.*, 2010, **132**, 3783.
- 19 W. Kosaka, J. Zhang, Y. Watanabe and H. Miyasaka, *Inorg. Chem.*, 2022, **61**, 12698.
- 20 S. Noro, Y. Meng, K. Suzuki, M. Sugiura, Y. Hijikata, J. Pirillo, X. Zheng, K. Takahashi and T. Nakamura, *Inorg. Chem.*, 2021, **60**, 4531.
- 21 Y. Sekine, W. Kosaka, H. Kano, C. Dou, T. Yokoyama and H. Miyasaka, *Dalton Trans.*, 2016, **45**, 7427.
- 22 H. Furukawa, F. Gándara, Y.-B. Zhang, J. Jiang, W. L. Queen, M. R. Hudson and O. M. Yaghi, *J. Am. Chem. Soc.*, 2014, **136**, 4369.
- 23 K. Fukuhara, S. Noro, K. Sugimoto, T. Akutagawa, K. Kubo and T. Nakamura, *Inorg. Chem.*, 2013, **52**, 5630.
- 24 M. Nakaya, Y. Kitagawa, S. Watanabe, R. Teramoto, I. Era, M. Nakano and J. Onoe, *Adv. Sustainable Syst.*, 2021, **5**, 2000156.
- 25 *CRC Handbook of Chemistry and Physics*, ed. David R. Lide, CRC Press, 88th edn, 2008.
- 26 A. Schneemann, Y. Takahashi, R. Rudolf, S. Noro and R. A. Fischer, *J. Mater. Chem. A*, 2016, **4**, 12963.
- 27 (a) A. Wang, X. Zheng, Y. Saito, A. Tateishi, Y. Huang, Y. Kamiya, H. Sato, A. Kondo, K. Takahashi, T. Nakamura and S. Noro, CCDC 2428329: Experimental Crystal Structure Determination, 2025, DOI: [10.5517/ccdc.csd.cc2mhw67](https://doi.org/10.5517/ccdc.csd.cc2mhw67); (b) A. Wang, X. Zheng, Y. Saito, A. Tateishi, Y. Huang, Y. Kamiya, H. Sato, A. Kondo, K. Takahashi, T. Nakamura and S. Noro, CCDC 2428330: Experimental Crystal Structure Determination, 2025, DOI: [10.5517/ccdc.csd.cc2mhw78](https://doi.org/10.5517/ccdc.csd.cc2mhw78).

