# Chemical Science



# **EDGE ARTICLE**

View Article Online
View Journal | View Issue



Cite this: Chem. Sci., 2025, 16, 16737

d All publication charges for this article have been paid for by the Royal Society of Chemistry

# Supramolecular conformational control of photolability in polymer networks crosslinked with a kinetically stable pseudo[1]rotaxane based on a coumarinylmethyl ester

Hiroshi Masai, (1)\*\* Naoki Niikura, Go M. Russell, Yutaro Kawano, Susumu Tsuda, (1)\*\* Tomohiro Iwai (1)\*\* and Jun Terao (1)\*\* Terao (1)\*\* Tomohiro Iwai (1)\*\* Alice teras (1)\*\* Tomohiro Iwai (1)\*\* Alice teras (1

Photolabile polymer materials have attracted considerable scientific and societal attention owing to the spatiotemporal control of light. However, easily photoreactive polymers are intrinsically unstable, leading to a tradeoff between their reactivity and stability. This article reports a proof-of-concept for reversible switching between photostable and photolabile states in polymer network materials enabled by the supramolecular transformation of the kinetically stable pseudo[1]rotaxane. A photocleavable coumarinylmethyl ester derivative, covalently linked with permethylated α-cyclodextrins, can be switched between insulated and uninsulated structures through conformational isomerization by altering the solvent polarity and heating. The hydrophobic environment of the cyclodextrin in the insulated structure partially inhibits polar solvation of the contact ion pair, an intermediate in the photocleavage reaction, thereby decreasing the photoreactivity of the coumarinylmethyl ester. Consequently, polymer network materials crosslinked with the insulated structure exhibit photostability, whereas the corresponding uninsulated structure is photolabile, demonstrating the reversible control of material photoreactivity via supramolecular conformational transformations.

Received 1st March 2025 Accepted 29th July 2025

DOI: 10.1039/d5sc01641j

rsc.li/chemical-science

#### Introduction

Light can induce macroscopic changes in photolabile materials such as deformation, softening, or liquefaction.<sup>1-4</sup> The remote adaptability and high spatiotemporal control of light facilitate the precise tuning of the shape and physical or chemical properties of materials. As a result, photolabile materials have been utilized in various applications, including photoprocessing, soft actuators, and drug delivery systems.<sup>5-7</sup> These materials can efficiently respond by cleaving the crosslinking points of the polymer network via external stimuli.8,9 Thus, photolabile polymers incorporating photocleavable crosslinkers, such as o-nitrobenzyl derivatives, coumarinylmethyl esters, and metal complexes, have been increasingly reported.9-14 Although these materials are typically designed to be highly sensitive to light for increased responsiveness, their rapid photoresponsiveness can also render them susceptible to light. This can lead to unintended decomposition under ordinary light conditions, restricting their long-term practical

The active control of the photoreactive and photostable states in polymer materials has recently gained increasing attention as a compatible methodology between photostability and photoreactivity. <sup>15-21</sup> For example, multistep and concerted reactions with light and additional stimuli in polymer materials have been explored for advanced photoresponsiveness beyond simple photodegradation. The new type of photoreactivity enables on-demand control of the photoreactivity and photostability of materials. <sup>22</sup> Our research group recently discovered the acid-induced photodegradability of insulated platinum acetylide complexes with permethylated  $\alpha$ -cyclodextrin (PM  $\alpha$ -CD) which show concerted reactivity with light and acid, leading to the development of polymer network materials with photoresponsiveness in the presence of acid. <sup>23,24</sup> The platinum complex demonstrated high photostability against acid owing

applications. Conversely, increasing the stability of materials to light can compromise their photoresponsiveness, presenting a trade-off between rapid photoreactivity and long-term photostability. However, overcoming these challenges through the on-demand active control of both the photoresponsive and photostable states could broaden the applications of these materials, rendering them suitable for long-term use under ordinary light conditions and enabling efficient photoreactivity under weak light conditions.

<sup>&</sup>lt;sup>a</sup>Department of Basic Science, Graduate School of Arts and Sciences, The University of Tokyo, 3-8-1, Komaba, Meguro-ku, Tokyo 153-8902, Japan. E-mail: cmasai.h@g.ecc.u-tokyo.ac.jp; cterao@g.ecc.u-tokyo.ac.jp

<sup>&</sup>lt;sup>b</sup>PRESTO, Japan Science and Technology Agency, 4-1-8, Honcho, Kawaguchi, Saitama 332-0012, Japan

Department of Chemistry, Osaka Dental University, 8-1 Kuzuhahanazonocho, Hirakata, Osaka 573-1121, Japan

to the steric effect of cyclodextrins. In contrast, when exposed to both light and acid, the complex exhibited cooperative cleavage reactivity. This unique property allows for novel reactivity in materials employing the complex as a network crosslinker.

Herein, a new molecular design for controlling the photodegradability and photostability of materials was developed through the supramolecular conformational isomerization of the pseudo[1]rotaxane structure within polymer networks. Supramolecular host-guest structures are widely used to stabilize guest molecules through the steric hindrance provided by the inner cavity of cyclic molecules. 25-29 In this study, a kinetically stable pseudo[1]rotaxane composed of photolabile guest moieties covalently linked with PM α-CD as a host was designed. The structure facilitated reversible switching between the insulated and uninsulated states through supramolecular conformational isomerization. 30-34 Furthermore, by introducing appropriate steric barriers to threading and dethreading in pseudo[1]rotaxanes, each isomer was kinetically stabled at room temperature, while intramolecular isomerization was triggered by altering the solvent polarity and heating.35 As the photolabile guest moiety, coumarinylmethyl ester derivatives were used, 10,36-40 which involves the formation of contact ion pairs (CIPs) upon light irradiation, as a proposed reaction mechanism of photodegradation (Fig. 1a).41-43 The CIP intermediates can be stabilized in polar solvents, which increases the photolabile reaction rate. In contrast, in the insulated structure, the inner cavity of PM  $\alpha$ -CD provides a hydrophobic environment around the coumarinylmethyl ester moiety, resulting in slower reaction rates due to the inhibition of the formation of ionic species in the hydrophobic cavity (Fig. 1b). Thus, this methodology offers photoreactive and photostable states for photolabile polymer materials that are reversible via conformational control. Although several supramolecular structures have been employed to control photoreactivity,44-47 the reversible intramolecular switching of photolability and its

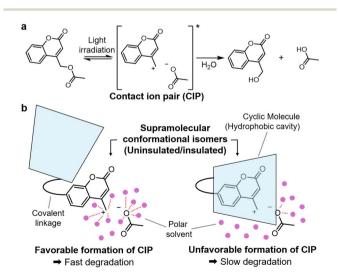


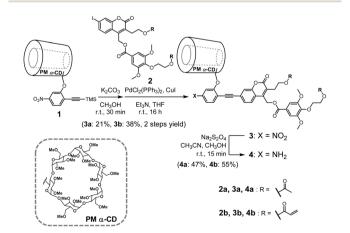
Fig. 1 (a) Photocleavage reaction of coumarinylmethyl ester derivatives via a CIP intermediate. (b) Conceptual diagram illustrating the control of the photolability using uninsulated and insulated structures (supramolecular conformational isomers).

application to the photoresponsiveness of polymer materials have not been reported. In this study, the photostability and photodegradability of polymer materials were actively controlled by the supramolecular isomerization of the kinetically stable pseudo[1]rotaxane comprising photolabile coumarinylmethyl ester derivatives covalently linked to PM  $\alpha\text{-CD}.$  This strategy allows the materials to transition between a photostable state and a photolabile state on-demand.

### Results and discussion

Coumarinylmethyl ester derivatives bearing PM  $\alpha$ -CD as the side chain (4a and 4b) were synthesized, as shown in Scheme 1.<sup>48</sup> A Sonogashira coupling reaction was conducted between an ethynylbenzene derivative bearing PM  $\alpha$ -CD (1) and a coumarinylmethyl ester bearing an iodo group (2), resulting in an uninsulated coumarinylmethyl ester derivative (3). The dimethoxybenzene moiety was introduced on 3 to monitor the photocleavage of the coumarinylmethyl ester using <sup>1</sup>H NMR spectroscopy because the hydrogen atoms on the dimethoxybenzene moiety exhibited characteristic chemical shifts after photodegradation ( $\sim$ 7.2 ppm). Subsequently, the nitro groups on 3a and 3b were converted into the corresponding amino groups using sodium hydrosulfite, yielding 4a to investigate the photoreactivity and 4b for use as a cross-linking reagent.

Compound **4a** underwent supramolecular conformational isomerization to its insulated counterpart (**4a**') by heating the solution in CH<sub>3</sub>OH-H<sub>2</sub>O (1/1, v/v) at 40 °C, using a hydrophilic solvent mixture (Fig. 2a). <sup>1</sup>H NMR analyses demonstrated the quantitative intramolecular transformation of **4a** to **4a**' within 2 h, as shown in Fig. 2b (a Detailed discussion is provided in the SI). In addition, treating the insulated structure (**4a**') in CHCl<sub>3</sub> at 60 °C for 10 h quantitatively converted it back to its uninsulated counterpart (**4a**), demonstrating the reversible transformation between the insulated and uninsulated structures (Fig. S2). Notably, the amino group at the axle terminal significantly increases the activation barrier for the threading/dethreading of the pseudo[1]rotaxane owing to steric hindrance.<sup>49</sup> As a result,



**Scheme 1** Synthesis route of the uninsulated coumarinylmethyl ester derivatives (4a and 4b).

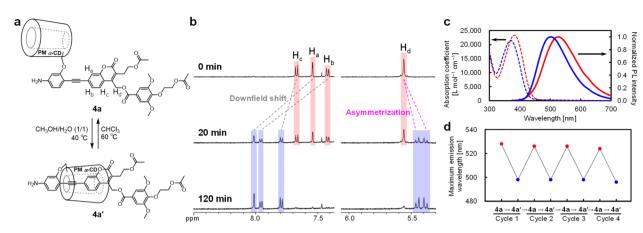


Fig. 2 (a) Intramolecular conversion between supramolecular conformational isomers (4a and 4a'). (b) Time-course of the  $^1H$  NMR spectra (500 MHz, CDCl<sub>3</sub>, r.t.) from 4a to 4a' under CH<sub>3</sub>OH-H<sub>2</sub>O (1/1, v/v) at 40 °C. (c) UV-vis absorption (dashed lines) and fluorescent (solid lines) spectra of 4a (red) and 4a' (blue) in CHCl<sub>3</sub>. (d) Variations in the maximum fluorescent wavelength under repetitive transformation cycles between the insulated (4a') and uninsulated (4a) structures (excitation at 365 nm in CHCl<sub>3</sub>).

the  $^1$ H NMR spectra indicated that 4a and 4a' remained stable at r.t. after 6 h, even in thermodynamically unfavorable solvents, including CDCl<sub>3</sub> for 4a' and CD<sub>3</sub>OD for 4a (Fig. S3 and S4). Therefore, the supramolecular conformational isomers (4a and 4a') were kinetically stable across a range of solvent polarities, enabling systematic investigation of insulation effects on photoreactivity by directly comparing insulated and uninsulated species in identical solvents.

The UV-vis absorption spectra of **4a** and **4a**′ in CHCl<sub>3</sub> exhibit a broad absorption band at 320–450 nm, with maximum wavelengths of 382 and 372 nm, respectively (Fig. 2c). Furthermore, the emission spectra of **4a** and **4a**′ in CHCl<sub>3</sub> under excitation at 365 nm exhibit maximum emission wavelengths at 525 and 501 nm, respectively, which is attributed to the donoracceptor type emission of the coumarinylmethyl ester

derivative. The blue-shift in the absorption and emission wavelengths of the insulated structure (4a') was attributed to the twisted  $\pi$ -conjugated system resulting from the threading with PM  $\alpha$ -CD, which decreased the effective conjugation length. The Furthermore, reversible conformational isomerizations between 4a and 4a' were achieved by heating solutions of 4a and 4a' in CH<sub>3</sub>OH-H<sub>2</sub>O (1/1, v/v) and CHCl<sub>3</sub>, respectively. Notably, the maximum emission wavelengths remained consistent even after four transformation cycles (Fig. 2d). These optical measurements indicate the successful synthesis of coumarinylmethyl ester-based  $\pi$ -systems with insulated and uninsulated structures capable of repeated switching.

The photodegradabilities of **4a** and **4a**′ were evaluated by tracing the <sup>1</sup>H NMR spectra under light irradiation (Fig. 3a). Coumarinylmethyl ester derivative solutions (**4a** or **4a**′) in

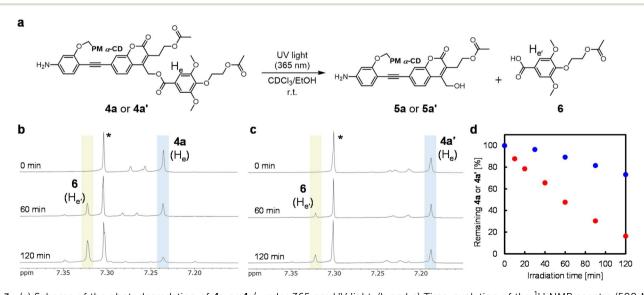


Fig. 3 (a) Scheme of the photodegradation of 4a or 4a' under 365 nm UV light. (b and c) Time-evolution of the  $^1$ H NMR spectra (500 MHz, CDCl<sub>3</sub>-EtOH (95/5, v/v), r.t.) of (b) 4a and (c) 4a' before and after photoirradiation with 365 nm UV light (480 mW cm<sup>-2</sup>). Asterisk: CHCl<sub>3</sub>. (d) Time-course of photodegradation of 4a (red) or 4a' (blue) analyzed by  $^1$ H NMR spectroscopy under 365 nm UV light irradiation.

This article is licensed under a Creative Commons Attribution 3.0 Unported Licence.

Open Access Article. Published on 19 August 2025. Downloaded on 9/25/2025 5:30:32 PM.

**Chemical Science Edge Article** 

CDCl<sub>3</sub>-EtOH (95/5, v/v) were exposed to 365 nm UV light (480 mW cm<sup>-2</sup>). Time-evolution of <sup>1</sup>H NMR spectra indicated that **4a** or 4a' was converted during photoirradiation, generating the syringic acid derivative (6) as the degradation product (Fig. 3b, c, S5, and S6). Additionally, the MS measurement after the photoreaction of 4a' supports the generation of coumarinylmethyl alcohol (5a or 5a') and a syringic acid derivative (6) (Fig. S10). The <sup>1</sup>H NMR analyses indicated that the degree of photodegradation for the uninsulated structure (4a) was approximately 70% after 90 min of exposure to 365 nm UV light (480 mW cm<sup>-2</sup>). In contrast, the photodegradation of the insulated structure (4a') was significantly reduced to approximately 20% (Fig. 3d). Thus, the insulation of 4a' effectively decreased the photodegradability of the coumarinylmethyl ester moiety, enhancing its photostability. Typical coumarinylmethyl esters exhibited increased photoreactivity in high-polarity solvents, indicating pronounced solvent dependence. 42 Hence, the observed difference in photolability between 4a and 4a' in CDCl3-EtOH was likely attributable to the low-polarity environment of the inner cavity within PM  $\alpha$ -CD.

To reveal the effects of the supramolecular structures, the photocleavage reactivities of 4a and 4a' were assessed using <sup>1</sup>H NMR in DMSO- $d_6$ , a high-polarity solvent. The photodegradation degree of the uninsulated molecule (4a) increased 2.4-fold after 10 min in DMSO-d<sub>6</sub> compared with that in the CDCl<sub>3</sub>-EtOH (95/5, v/v) mixture (Fig. S7 and S8). However, the photodegradation degree of 4a' in DMSO- $d_6$  was similar to that of 4a (Fig. S9), despite its insulated structure. This result is

possibly due to the strong polar solvation in DMSO- $d_6$ , which may have surpassed the insulating effect on the coumarinylmethyl ester moiety. The CPK model of the insulated structure shown in Fig. S16 was obtained through computational calculations, revealing that the coumarinylmethyl ester was primarily located in the inner to peripheral region of PM  $\alpha$ -CD, while the reactive site was slightly exposed to the external environment. Consequently, in the relatively low-polarity solvent (CDCl<sub>3</sub>-EtOH), the limited exposure was insufficient to stabilize the CIP intermediate, and the PM α-CD exerted an insulating effect. In contrast, in DMSO- $d_6$ , the high polarity solvent effectively stabilized even partially exposed ion-pair intermediates via solvation. This solvation effect would override the stabilizing effect of the macrocycle, resulting in similar photoreactivities of 4a and 4a' in DMSO-d<sub>6</sub>. Although the stabilizing effects depended on the solvent system, this study reveals that isomeric supramolecular protection can govern intramolecular photoreactivity by modulating solvation dynamics at the reactive site. This highlights the potential of supramolecular conformational isomers to modulate the intramolecular photolability of polymer materials, suggesting a new strategy for controlling their durability and sensitivity under light exposure.

Polymer network materials incorporating coumarinylmethyl ester derivatives were synthesized via radical polymerization using N-isopropylacrylamide (NIPAM) as a monomer (1 eq.), 4b as a photolabile crosslinker (0.001 or 0.0015 eq.), and N,N'-methylenebisacrylamide (0 or 0.001 eq.) as a photostable

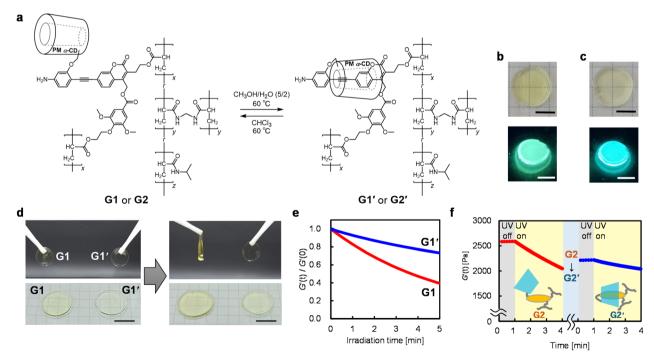


Fig. 4 (a) Reversible transformations of the uninsulated and insulated gels. (b and c) Photographic images of (b) the G1 and (c) G1' gels under white (top) and 365 nm UV light (bottom) irradiation (x/y/z = 0.15/0/100, scale bar: 5 mm). (d) Photographic images of the uninsulated (G1) and insulated (G1') gels before (left) and after (right) photoirradiation (365 nm, 116 mW cm<sup>-2</sup>, scale bar: 10 mm). (e) Time-evolution of the storage moduli (G') of photoirradiated (365 nm, 33 mW cm<sup>-2</sup>) G1 and G1' in CHCl<sub>3</sub>-EtOH (95/5, v/v). (f) Time-evolution of the G' of G2 (x/y/z = 0.1/0.1/100) and the gel subsequently transformed to an insulated structure (G2') during photoirradiation (365 nm, 33 mW cm<sup>-2</sup>) measured in CHCl<sub>3</sub>-EtOH (95/5, v/v).

crosslinker. The radical polymerization was initiated by 2,2′-azobis(2,4-dimethylvaleronitrile) (0.001 eq.) at 60 °C for 18 h in DMSO. The resulting network material (G1, x/y/z = 0.15/0/100 in Fig. 4a), which featured an uninsulated coumarinylmethyl ester moiety, yielded a transparent and elastic disc-shaped gel (Fig. 4b). G1, which is based on a poly(NIPAM), could retain solvents of various polarities. The swelling ratios, which are calculated based on the swollen mass to the dry mass of the gel, were 37, 47, and 12 for different polarity solvents (CHCl<sub>3</sub>, CHCl<sub>3</sub>–EtOH (95/5, v/v), and CH<sub>3</sub>OH–H<sub>2</sub>O (5/2, v/v), respectively). The result demonstrates the amphiphilic nature of the poly(NIPAM) network, which enhances its broad capability of retaining the polarity of solvents.

The uninsulated moiety in G1 was converted to its insulated counterpart (G1') through supramolecular conformational isomerization by soaking it in a hydrophilic CH<sub>3</sub>OH-H<sub>2</sub>O (5/2, v/ v) solution at 60 °C (Fig. 4a). The luminescence spectrum of the resulting gel exhibited a peak emission wavelength of 498 nm, which was blue-shifted by approximately 20 nm, as compared with that of the original gel, G1 (518 nm) (Fig. 4b, c, S11a, and S11e). This shift in the emission wavelength upon insulation corresponds with the observed trends in the conversion between compounds 4a and 4a'. Furthermore, the emission wavelengths of G1 and G1' barely changed (~3 nm) after 6 h even in high (CH<sub>3</sub>OH-H<sub>2</sub>O (5/2, v/v)) and low (CHCl<sub>3</sub>) polarity solvents, which are thermodynamically unfavorable solvents for the uninsulated and insulated structures, respectively (Fig. S11c, S11d, and S11f). The kinetic stability of the insulated structure was maintained even in the gel state (G1'). In contrast, under CHCl<sub>3</sub> at 60 °C, the insulated structure of G1' reverted to the uninsulated form, G1, and the emission spectrum shifted back to its original state (Fig. S11b and S11e). This reversible transformation between the insulated and uninsulated structures, along with their kinetic stabilization, was also demonstrated in the network material.

The macroscopic properties of the gel materials containing the insulated and uninsulated crosslinkers changed drastically due to the cleavage of the crosslinking points. **G1** was degraded under photoirradiation and lost its original shape, whereas the corresponding insulated gel (**G1**′) maintained its original shape under the same irradiation conditions (365 nm, 116 mW cm $^{-2}$ , 20 min) (Fig. 4d). Even though the molar ratio of the photolabile crosslinker to the monomer was small (~1/1000 eq.), the photoreactive crosslinker significantly influenced the macroscopic properties of the gel. Hence, the photoreactivity of the insulated and uninsulated coumarinylmethyl ester derivatives was reflected in the macroscopic photoresponsiveness of the polymer network material.

The photodegradabilities of the gels were quantitatively evaluated by tracing their dynamic viscoelasticity under 365 nm UV light irradiation. This process decreases the storage modulus (G') of the gel, which is proportional to the crosslinking density. The G' of the uninsulated gel (G1) decreased upon irradiation with 365 nm UV light (33 mW cm<sup>-2</sup>) using CHCl<sub>3</sub>–EtOH (95/5, v/v) as the retained solvent (Fig. 4e). Conversely, the G' of the insulated gel (G1') did not relatively decrease. These results indicate that the kinetically stable

pseudo[1]rotaxane in the coumarinylmethyl ester enhances the photostability of the gels, which is consistent with the trends observed for 4a and 4a'. Moreover, the difference in the photodegradability between G1 and G1' was dependent on the solvent polarity of the gels; The degradation rates of G1 and G1' in a high-polarity solvent (DMSO) were approximately consistent (Fig. S12). The macroscopic photolabilities of G1 and G1' successfully reflected the nanoscale molecular reactivities of 4a and 4a'.

Finally, the photoreactivity of the network can be switched after partial degradation by inducing a conformational transformation of the pseudo[1]rotaxane structure. G2 (x/y/z = 0.1/z)0.1/100), a polymer network material, was first irradiated with 365 nm UV light (33 mW cm<sup>-2</sup>) for 3 min in the retaining solvent CHCl<sub>3</sub>-EtOH (95/5, v/v). This cleaved a portion of the coumarinylmethyl-ester cross-links, lowering the normalized G' by 21% (Fig. 4f). The partially photodegraded G2 was then insulated by solvent exchange to CH<sub>3</sub>OH-H<sub>2</sub>O (5/2, v/v) and heating to 60 °C, triggering isomerization to the insulated structure (G2'). After replacing the retaining solvent with CHCl<sub>3</sub>-EtOH (95/5, v/v), the material was re-irradiated under the same UV conditions, resulting in a slight decrease in normalized G' by only 8% after 3 min. These results demonstrate that the conformational change of the kinetically stable pseudo[1] rotaxane converts the network from a photolabile to a photostable state, as intended. Accordingly, the internal molecular switch enables active control over the material's resistance to photodegradation.

#### Conclusions

In summary, a photolabile coumarinylmethyl ester derivative was insulated by the kinetically stable pseudo[1]rotaxane using PM α-CD. This design enabled control of the photolability via a quantitative and reversible intramolecular transformation between supramolecular conformational isomers. The insulated structure enhanced photostabilization by creating a hydrophobic environment around the photolabile moiety, while the uninsulated counterpart exhibited efficient photodegradability. Furthermore, the adaptable photodegradability can be incorporated into polymers as crosslinking points, creating the first instance of photolabile polymer networks with reversible intramolecular switching of photoreactivity through supramolecular conformational isomerization. An examination of the photoreactivity of both the insulated and uninsulated molecules revealed that the polymer network with the insulated crosslinker remained stable under photoexcitation, whereas the uninsulated network exhibited photodegradability. Moreover, the photostability and photodegradability could be switched by the supramolecular conformational transformation of the kinetically stable pseudo[1]rotaxanes in the network materials. To advance this strategy from proof-of-concept to real-world application, several challenges remain, including the design of molecules suitable for efficient synthesis on an industrial scale and the development of simplified switching protocols compatible with practical materials. Despite these challenges, the present work establishes a new platform for on-demand

control of material photolability, reconciling the long-standing trade-off between photostability and photoreactivity.

#### **Author contributions**

H. M. and J. T. conceptualized the study, acquired funding, and administered the project. H. M., T. I., and J. T. supervised the project. G. M. R. conducted preliminary investigations. N. N., G. M. R., and Y. K. developed the methodologies and undertook the investigations, including the experiments and analyses. T. S. undertook the synthesis experiments. H. M. prepared the original manuscript draft, and all authors contributed to reviewing and editing the submitted version.

#### Conflicts of interest

The authors declare no competing interests.

# Data availability

The SI includes experimental details and data from the NMR, MS, optical, and viscoelastic analyses. See DOI: https://doi.org/10.1039/d5sc01641j.

# Acknowledgements

This research was supported by JST PRESTO (grant number JPMJPR21N8), JST CREST (grant number JPMJCR19I2), JSPS KAKENHI (grant numbers 21K05181, 21K18948, 22H02060, and 25K01827), the Toshiaki Ogasawara Memorial Foundation and the Iketani Science and Technology Foundation.

#### Notes and references

- 1 M. Gernhardt, V. X. Truong and C. Barner-Kowollik, Adv. Mater., 2022, 34, 2203474.
- 2 W.-J. Li, W.-T. Xu, X.-Q. Wang, Y. Jiang, Y. Zhu, D.-Y. Zhang, X.-Q. Xu, L.-R. Hu, W. Wang and H.-B. Yang, *J. Am. Chem. Soc.*, 2023, 145, 14498–14509.
- 3 T. E. Brown and K. S. Anseth, *Chem. Soc. Rev.*, 2017, 46, 6532–6552
- 4 L. Li, J. M. Scheiger and P. A. Levkin, *Adv. Mater.*, 2019, 31, 1807333.
- 5 C. Zhang, Z. Zeng, D. Cui, S. He, Y. Jiang, J. Li, J. Huang and K. Pu, *Nat. Commun.*, 2021, 12, 2934.
- 6 E. Fuentes, Y. Gabaldón, M. Collado, S. Dhiman, J. A. Berrocal, S. Pujals and L. Albertazzi, *J. Am. Chem. Soc.*, 2022, 144, 21196–21205.
- 7 C. Yang, F. W. DelRio, H. Ma, A. R. Killaars, L. P. Basta, K. A. Kyburz and K. S. Anseth, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, 113, E4439–E4445.
- P. J. LeValley, R. Neelarapu, B. P. Sutherland, S. Dasgupta,
   C. J. Kloxin and A. M. Kloxin, *J. Am. Chem. Soc.*, 2020, 142, 4671–4679
- 9 A. M. Kloxin, A. M. Kasko, C. N. Salinas and K. S. Anseth, *Science*, 2009, 324, 59–63.

- 10 M. A. Azagarsamy, D. D. McKinnon, D. L. Alge and K. S. Anseth, *ACS Macro Lett.*, 2014, 3, 515–519.
- S. Theis, A. Iturmendi, C. Gorsche, M. Orthofer, M. Lunzer,
   S. Baudis, A. Ovsianikov, R. Liska, U. Monkowius and
   I. Teasdale, *Angew. Chem., Int. Ed.*, 2017, 56, 15857–15860.
- 12 M. A. Ayer, Y. C. Simon and C. Weder, *Macromolecules*, 2016, 49, 2917–2927.
- 13 N. J. Oldenhuis, K. P. Qin, S. Wang, H. Z. Ye, E. A. Alt, A. P. Willard, T. Van Voorhis, S. L. Craig and J. A. Johnson, Angew. Chem., Int. Ed., 2020, 59, 2784–2792.
- 14 S. Honda and T. Toyota, Nat. Commun., 2017, 8, 502.
- 15 B. A. Badeau, M. P. Comerford, C. K. Arakawa, J. A. Shadish and C. A. DeForest, *Nat. Chem.*, 2018, **10**, 251–258.
- 16 J. Lai, L. Abune, N. Zhao and Y. Wang, Angew. Chem., Int. Ed., 2019, 58, 2820–2825.
- 17 Q. Huang, T. Liu, C. Bao, Q. Lin, M. Ma and L. Zhu, *J. Mater. Chem. B*, 2014, **2**, 3333–3339.
- 18 H. Makino, T. Nishikawa and M. Ouchi, *Macromolecules*, 2023, **56**, 8776–8783.
- 19 H. Masai, T. Nakagawa and J. Terao, *Polym. J.*, 2024, **56**, 297–307.
- 20 K. Kalayci, H. Frisch, V. X. Truong and C. Barner-Kowollik, *Nat. Commun.*, 2020, **11**, 4193.
- 21 Y. Kawano, H. Masai, T. Tsubokawa, D. Yokogawa, T. Iwai and J. Terao, *Adv. Mater.*, 2025, 37, 2412544.
- 22 T. Inui, E. Sato and A. Matsumoto, ACS Appl. Mater. Interfaces, 2012, 4, 2124–2132.
- 23 T. Kaneko, G. M. Russell, Y. Kawano, H. Masai and J. Terao, Angew. Chem., Int. Ed., 2023, 62, e202305374.
- 24 H. Masai, Bull. Chem. Soc. Jpn., 2023, 96, 1196-1205.
- 25 M. J. Frampton and H. L. Anderson, Angew. Chem., Int. Ed., 2007, 46, 1028–1064.
- 26 C. Pan, C. Zhao, M. Takeuchi and K. Sugiyasu, *Chem.-Asian J.*, 2015, **10**, 1820–1835.
- 27 M. Cirulli, A. Kaur, J. E. M. Lewis, Z. Zhang, J. A. Kitchen, S. M. Goldup and M. M. Roessler, *J. Am. Chem. Soc.*, 2019, 141, 879–889.
- 28 K. Kano, H. Kitagishi, M. Kodera and S. Hirota, *Angew. Chem.*, *Int. Ed.*, 2005, **44**, 435–438.
- 29 T. Hosomi, R. Harada, H. Masai, T. Fujihara, Y. Tsuji and J. Terao, *Chem. Commun.*, 2018, **54**, 2487–2490.
- 30 K. Hiratani, M. Kaneyama, Y. Nagawa, E. Koyama and M. Kanesato, *J. Am. Chem. Soc.*, 2004, **126**, 13568–13569.
- 31 A. Miyawaki, P. Kuad, Y. Takashima, H. Yamaguchi and A. Harada, *J. Am. Chem. Soc.*, 2008, **130**, 17062–17069.
- 32 T. Takata, ACS Cent. Sci., 2020, 6, 129-143.
- 33 R. Zhang, C. Wang, R. Long, T. Chen, C. Yan and Y. Yao, *Front. Chem.*, 2019, 7, 508.
- 34 M. Xue, Y. Yang, X. Chi, X. Yan and F. Huang, *Chem. Rev.*, 2015, **115**, 7398–7501.
- 35 H. Masai, M. Liu, Y. Tachibana, S. Tsuda and J. Terao, *J. Org. Chem.*, 2020, **85**, 3082–3091.
- 36 T. Furuta, S. S. H. Wang, J. L. Dantzker, T. M. Dore, W. J. Bybee, E. M. Callaway, W. Denk and R. Y. Tsien, *Proc. Natl. Acad. Sci. U. S. A.*, 1999, 96, 1193–1200.

37 P. Klán, T. Šolomek, C. G. Bochet, A. Blanc, R. Givens, M. Rubina, V. Popik, A. Kostikov and J. Wirz, Chem. Rev.,

**Edge Article** 

- 2013, 113, 119-191.
- 38 W. Fan, X. Tong, Q. Yan, S. Fu and Y. Zhao, Chem. Commun., 2014, 50, 13492-13494.
- 39 Q. Lin, L. Yang, Z. Wang, Y. Hua, D. Zhang, B. Bao, C. Bao, X. Gong and L. Zhu, Angew. Chem., Int. Ed., 2018, 57, 3722-
- 40 R. Schmidt, D. Geissler, V. Hagen and J. Bendig, J. Phys. Chem. A, 2007, 111, 5768-5774.
- 41 A. M. Schulte, G. Alachouzos, W. Szymański and B. L. Feringa, J. Am. Chem. Soc., 2022, 144, 12421-12430.
- 42 A. M. Schulte, G. Alachouzos, W. Szymanski and B. L. Feringa, Chem. Sci., 2024, 15, 2062-2073.

- 43 H. D. Nguyen and M. Abe, J. Am. Chem. Soc., 2024, 146, 10993-11001.
- 44 V. Ramamurthy, Acc. Chem. Res., 2015, 48, 2904-2917.
- 45 V. Ramamurthy and S. Gupta, Chem. Soc. Rev., 2015, 44, 119-
- 46 S. Mei, Q. Ou, X. Tang, J. F. Xu and X. Zhang, Org. Lett., 2023, 25, 5291-5296.
- 47 D. Liu, Y. Fan, M. Liu, Q. Ge, R. Gao and H. Cong, Org. Lett., 2024, **26**, 3896-3900.
- 48 G. M. Russell, H. Masai and J. Terao, Phys. Chem. Chem. Phys., 2022, 24, 15195-15200.
- 49 H. Masai, J. Terao, T. Fujihara and Y. Tsuji, Chem. Eur. J., 2016, 22, 6624-6630.
- 50 H. Masai, T. Fujihara, Y. Tsuji and J. Terao, Chem. Eur. J., 2017, 23, 15073-15079.