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In situ transmission electron microscopy characterization and manipulation of the morphology, composition and phase evolution of nanomaterials under microenvironmental conditions

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Nanomaterials possess a broad range of applications in areas such as catalysis, energy, and biomedicine because of their unique properties. However, from the perspective of materials synthesis, there are numerous challenges in the controllable preparation of nanomaterials. These include the control of their size, morphology, crystal structure, and surface properties, which are essential for their performance in specific applications. The fundamental cause of these issues is the limitation in the real-time observation of the growth process of nanomaterials. *In situ* transmission electron microscopy (TEM), on the other hand, overcomes the limitations of traditional *in situ* testing techniques. It enables the real-time observation and analysis of the dynamic structural evolution during the growth of nanomaterials at the atomic scale. This contributes to a profound understanding of the nucleation and growth mechanisms of nanomaterials and facilitates the controlled preparation of nanomaterials. This review centers on the utilization of *in situ* TEM to observe and study the complex dynamic processes of zero-, one-, and two-dimensional nanomaterial growth and evolution in different environments (liquid, gas, and solid phases) at the atomic scale. This is of great significance for the design and preparation of nanomaterials with

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specific properties. The proposed future development of *in situ* TEM, in combination with advanced data analysis and integration with other techniques, holds great potential for the further advancement of nanotechnology and its applications.

1 Introduction

Nanomaterials, defined by their size typically ranging from 1 to 100 nanometers, have ignited a revolution in the field of materials science due to their unique properties and broad spectrum of applications. These materials exhibit distinctive characteristics that are not observed in their bulk counterparts, primarily due to their high surface-to-volume ratios, quantum confinement effects, and the ability to manipulate their structures at the atomic level. Nanomaterials have extremely extensive application prospects in the domains of catalysis,[1,2] energy,[3] and biomedicine.[4,5] Despite their immense potential, the development and application of nanomaterials also present challenges. These include issues related to their synthesis, stability, toxicity, and environmental impact. The controlled fabrication of nanomaterials, or the precise control over their size, shape, crystal structure, and surface properties, is a complex and challenging process that is critical for tailoring their behaviors and optimizing their performance in various applications.[6–8] The quest for precision in crafting nanomaterials with tailored properties is hindered by a complex interplay of factors that govern their formation, growth, and resultant characteristics. One of the primary obstacles in the controlled synthesis of nanomaterials is the deep understanding and manipulation of nucleation and growth mechanisms at the atomic scale.[9,10] As demonstrated in the accompanying documents, the journey from monomers to stable nanocrystals is fraught with complexity. Classical and non-classical nucleation theories attempt to explain the aggregation of intermediates into crystalline structures, although the reality of atomic migration dynamics, interfacial evolution, and structural transformation during synthesis often deviates from these theoretical predictions.[11,12] The uniformity and scalability of nanomaterial synthesis are also significant challenges. The high surface-to-volume ratio of nanomaterials, which is a source of their unique properties, also introduces variability in their synthesis. Techniques such as wet chemical synthesis and solid-state reactions often yield nanomaterials with a broad size distribution and morphological diversity, which can compromise their performance in specific applications. Moreover, the influence of environmental factors during synthesis, such as temperature, pressure, and the presence of surfactants or solvents, adds another layer of complexity. These factors can significantly affect the crystallinity, phase, and surface properties of nanomaterials, leading to a lack of reproducibility in their synthesis. The stability of nanomaterials during synthesis and under various conditions is a critical concern. Phase transformations and structural changes under different stimuli, such as thermal, mechanical, or chemical influences, can alter the intended properties of nanomaterials. For instance, the thermal stability of magnetic nanoparticles[13] is crucial for their application in high-temperature environments, and any phase change could render them ineffective.[14,15] In conclusion, the controlled synthesis of nanomaterials is a multifaceted challenge that requires a profound understanding of atomic-scale processes, mastery over environmental influences, ensuring stability and safety, and the development of scalable and environmentally friendly methods. Overcoming these challenges is crucial for the advancement of nanotechnology and the realization of its vast potential in diverse fields of application.

In the quest to unravel the intricate processes underlying the formation and evolution of nanomaterials, *in situ* transmission electron microscopy (TEM) has emerged as a transformative tool, providing a platform for real-time observation and manipulation of nanostructures with atomic accuracy.[16] The journey of *in situ* TEM began with the desire to transcend the

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Shujiang Ding

limitations of traditional *ex situ* characterization techniques, which often fell short in capturing the dynamic nature of material synthesis and phase transformations. *In situ* TEM has evolved to fill this void, offering a suite of capabilities that allow researchers to peer into the heart of nanomaterial formation processes. With its ability to operate under a variety of conditions, including high temperatures, pressures, and in the presence of various chemical environments,[17] *in situ* TEM has become an indispensable ally in the quest to understand and control material properties at the fundamental level.[18,19] The importance of *in situ* TEM in nanomaterial preparation cannot be overstated. It has facilitated the visualization of nucleation events, the tracking of growth pathways, and the exploration of structural dynamics in real time.[20] This has been particularly crucial in advancing our understanding of phenomena such as Ostwald ripening, phase separation, and defect evolution, which are pivotal in determining the final properties of nanomaterials.[21–23] Furthermore, *in situ* TEM has been instrumental in elucidating the mechanisms of size-dependent phase transformations and the role of surface energy in stabilizing metastable phases. One of the key strengths of *in situ* TEM lies in its multimodal approach, which integrates imaging with spectroscopic techniques such as energy dispersive X-ray spectroscopy (EDS) and electron energy loss spectroscopy (EELS). This synergy allows for comprehensive characterization of nanomaterials, capturing not only their morphology but also their chemical composition and electronic structure. The implementation of aberration-corrected lenses and the development of advanced imaging modalities, such as high-angle annular dark field (HAADF), scanning TEM (STEM) and electron tomography, have further enhanced the spatial resolution and analytical prowess of *in situ* TEM. The impact of *in situ* TEM extends beyond academic research. It has played a pivotal role in the development of new materials for energy storage, catalysis, electronics, and medicine. For instance, in the realm of catalysis, *in situ* TEM has been used to study the active sites of nanoparticles under reaction conditions, providing insights into their catalytic mechanisms and enabling the design of more efficient catalysts.[24,25] Similarly, in the field of electronics, the technique has been harnessed to investigate the conduction properties of nanomaterials and their response to electrical stimuli, which is vital for the development of nanoscale devices.[26,27] As we stand on the precipice of new discoveries, the future of *in situ* TEM holds promise for even greater advancements. The integration of machine learning algorithms and artificial intelligence is set to enhance data analysis and automate the identification of complex structural transformations. Moreover, the ongoing miniaturization of TEM components and the development of more sensitive detectors will further push the limits of spatial and temporal resolution, enabling the capture of fleeting events and transient states in nanomaterial systems. The development of *in situ* TEM has been a monumental achievement in the field of nanoscience, providing a powerful means to study and control the synthesis of nanomaterials. Its ability to offer insights into the atomic-scale processes that govern material properties has not only enriched our fundamental understanding but also paved the

way for innovative applications across diverse industries.[28] As we continue to push the boundaries of this technology, the potential for transformative breakthroughs in material design and engineering remains limitless.

In this review, we focused on the application of *in situ* TEM in analyzing the morphology, composition, and phase evolution of nanomaterials under microenvironmental conditions (Fig. 1). It highlights the challenges in controlling the synthesis of nanomaterials, such as nucleation and growth mechanisms, environmental influences, and stability issues. *In situ* TEM is presented as a transformative tool that allows real-time observation and manipulation of nanostructures with atomic precision, overcoming the limitations of traditional *ex situ* techniques. The review categorizes *in situ* TEM methodologies into several types, including heating chips, gas-phase cells, and liquid cells, each serving specific roles in nanomaterial synthesis. It discusses the insights gained from *in situ* TEM in understanding the nucleation and growth of nanocrystals, the formation of 0D, 1D, and 2D nanomaterials, and the effects of electron-beam interactions. The article also addresses the challenges associated with *in situ* TEM, such as replicating realistic synthesis conditions, achieving high temporal and spatial resolution, managing electron beam interactions, and integrating with other analytical techniques. Despite these challenges, the review emphasizes the significant contributions of *in situ* TEM to advancing our understanding of nanomaterial synthesis and its potential for future breakthroughs in material design and engineering.

2 Classifications of *in situ* TEM for nanomaterials synthesis

In situ TEM methodologies possess the capability to monitor the developmental stages of a particular system through the establishment and activation of its external conditions. Integral to *in situ* TEM are the application of external triggers and the ability to perform real-time monitoring. The former is facilitated by a range of specialized TEM holders, while the latter benefits from the implementation of advanced, rapid recording systems, which are not the focus of this review. To date, the exploration of TEM holders for nanomaterial synthesis has led to the identification of five types: the *in situ* heating chip, the electrochemical liquid cell,[33–35] the graphene liquid cell,[36–38] the gas-phase cell,[39,40] and the environmental TEM,[41,42] each playing a significant role in this field (Fig. 2).

2.1 *In situ* TEM thermal engineering—heating chip

The "heating chip" is a specialized tool utilized in the field of materials science, particularly for the *in situ* study of nanomaterials. It allows for the precise control and application of heat to samples within a TEM, enabling researchers to observe the dynamic evolution of the material structure and chemistry under elevated temperatures. Classifications of heating chips typically focus on the type of heating element and the method of temperature control.[47–49] For instance, there are furnace-type heating holders, which use a resistive heating element to heat
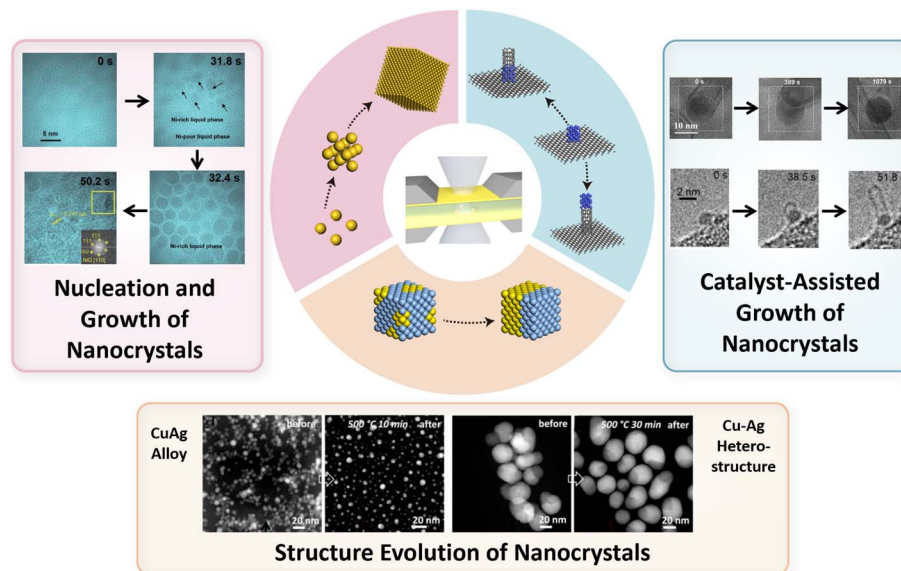
Fig. 1 Schematic illustration of key topics in this review. Reproduced with permission.[29–32] Copyright 2022, Tsinghua University Press. Copyright 2008, American Chemical Society. Copyright 2020, American Chemical Society. Copyright 2019, American Chemical Society.

the entire sample, and microelectromechanical system (MEMS)-based heating holders, which offer localized heating through a nanopatterned metal element.[50] MEMS-based heating chips are often pre-calibrated for temperature control but may require verification under specific experimental conditions due to probable electron-beam-induced heating effects.

In the synthesis of nanomaterials, heating chips play a crucial role by facilitating the study of thermally induced transformations. They allow researchers to monitor processes such as phase transitions, crystal growth, and structural changes in real time. The localized heating provided by MEMS-based chips is particularly beneficial for studying
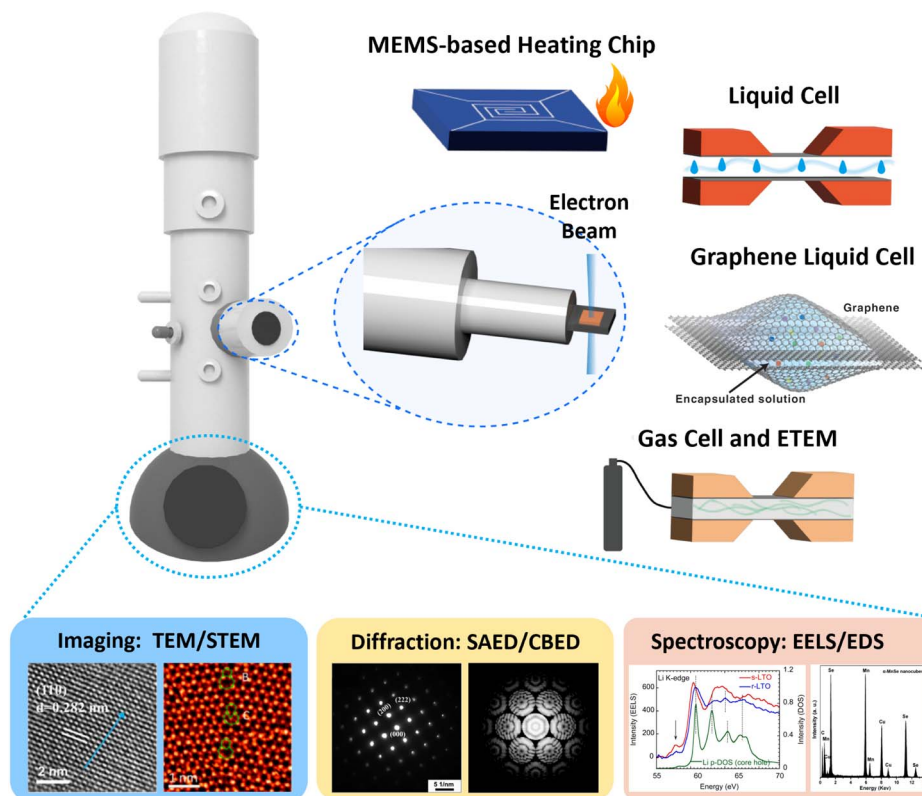

Fig. 2 Schematics of TEM and *in situ* chips and cells fabricated by MEMS. Reproduced with permission.[43–46] Copyright 2013, Elsevier. Copyright 2024, Elsevier. Copyright 2016, American Chemical Society. Copyright 2019, American Chemical Society.

nanostructured and focused ion beam samples, as it minimizes thermal drift and enables rapid temperature stabilization. Moreover, heating chips can be integrated with other *in situ* TEM techniques, such as gas or liquid environments, to simulate realistic conditions for the material under study. This integration provides a comprehensive understanding of the material's behavior under various stimuli, which is vital for the rational design and optimization of nanomaterials for energy-related applications. Heating chips are indispensable for *in situ* TEM studies, providing a controlled thermal environment to observe and analyze the structural and chemical evolution of nanomaterials, thereby contributing significantly to the advancement of materials science and technology.[51]

2.2 *In situ* TEM gas environment study—gas cell and environmental TEM

In situ TEM for gas environments is primarily classified into two types: environmental TEM (ETEM)[41] and windowed holders.[18] ETEMs feature advanced pumping systems that maintain a high vacuum in the microscope column while allowing gas to be introduced around the sample, thus simulating realistic reaction conditions. They have progressed from early models with limited resolution to modern versions capable of atomic-scale imaging. Windowed holders (gas cell), in contrast, are designed to contain both the material and gas within a sealed reactor that can be heated, enabling studies under controlled gas and temperature conditions. These holders are compatible with conventional TEM instruments and allow for experiments under high pressures, making them versatile for various catalytic studies.[52] In the context of nanomaterial synthesis, *in situ* TEM plays a vital role. It has been used to monitor the transformation of metal particles into single atoms, a process that significantly enhances catalytic efficiency. For example, the evolution of subnanometric metal species within specific spatial confinements has been studied under different gas atmospheres, revealing the dynamics of sintering and redispersion. Additionally, *in situ* TEM has been instrumental in visualizing the diffusion of atoms within matrices, leading to the formation of individual atoms and clusters, which is essential for the development of highly dispersed catalysts.

2.3 *In situ* TEM liquid environment study—electrochemical liquid cell and graphene liquid cell

Liquid cell technology has revolutionized the field of nanomaterial synthesis by enabling *in situ* observations under TEM. These cells are designed to contain liquid samples within a high-vacuum TEM environment, allowing researchers to directly visualize dynamic processes such as nanoparticle growth, transformation, and motion at the nanoscale. There are primarily two types of liquid cells: microfabricated silicon cells[33,53] and graphene liquid cells.[38] Microfabricated silicon cells are made from silicon wafers and feature electron-beam-transparent windows, typically composed of thin Si_3N_4 membranes, which allow for high-resolution imaging. These cells can be static or flow-type, with the latter enabling controlled introduction of reactants for reactions requiring precise temporal and spatial control. Graphene liquid cells, on the other hand, use graphene sheets as the window material, offering even higher spatial resolution due to the thinness and electron transparency of graphene.[54]

In the context of nanomaterial synthesis, liquid cells have been instrumental in studying various phenomena. They have been used to observe the nucleation and growth of nanoparticles, including metal nanoparticles, through processes such as electrochemical reduction.[55] The real-time imaging capabilities of liquid cells have shed light on the mechanisms of nanoparticle formation, revealing insights into the role of pre-nucleation intermediates and the dynamics of aggregative growth.[56] Furthermore, liquid cells have facilitated the investigation of nanoparticle transformations, such as galvanic replacement reactions and etching processes, which are crucial for synthesizing nanoparticles with complex structures and desired properties. The ability to control the chemical environment within liquid cells has also been vital for studying the effects of various factors on nanoparticle synthesis, such as the concentration of precursors, the presence of stabilizing agents, and the influence of the electron beam itself.[17] By manipulating these parameters, researchers can gain a deeper understanding of the underlying chemistry and optimize the synthesis of nanomaterials for various applications. Liquid cell technology has expanded the horizons of nanomaterial research by providing a platform for *in situ* TEM studies, offering unprecedented insights into the formation, transformation, and behavior of nanoparticles in liquid media. This technology has become an indispensable tool for scientists in the fields of materials science, chemistry, and nanotechnology.

3 Nucleation and growth of nanocrystals

3.1 Nucleation of nanocrystals

Over the past few years, there has been a consistent rise in the utilization of functional nanocrystals across different sectors, with a notable increase in energy, catalysis, and biomedicine applications.[1,3] This surge is primarily attributed to the distinctive structural characteristics of nanocrystals, such as their extensive specific surface area, one of a kind surface, and electron configurations, which hold promise for these applications. Nonetheless, crafting nanoparticles with the desired structures and attributes poses considerable difficulties. To advance further, a variety of theories regarding the nucleation and development of nanocrystals have been put forward, encompassing both classical and non-classical nucleation concepts, as well as growth mechanisms like ripening, clustering, and merging.[12] Despite these theoretical advancements, the underlying atomic movements, regulatory elements, and propelling forces of these phenomena are not yet fully understood.[21] In this context, the use of *in situ* TEM is emerging as an indispensable instrument for scientific inquiry, enhancing our comprehension of the mechanisms behind nanocrystal development. The insights gained from these studies are instrumental in the deliberate alteration of the structure and form of

nanocrystals, ultimately guiding the production of high-caliber nanocrystals tailored for diverse applications.

Classical nucleation theory is a fundamental concept in understanding the formation of nanocrystals.[57] The classical LaMer model (Fig. 3A)[58] is an extension of classical nucleation theory. It describes the process by which a new thermodynamic phase forms through the aggregation of monomer units. This theory outlines the process by which nanoparticles nucleate and

grow in a solution phase. Here is a summary of the key points related to classical nucleation theory:

3.1.1 Nucleation process. The nucleation of nanoparticles is described as a two-step process involving the formation of stable nuclei from supersaturated precursor species, followed by their growth into larger particles. The nucleation event is considered a critical phase transition from a supersaturated solution to a solid phase.
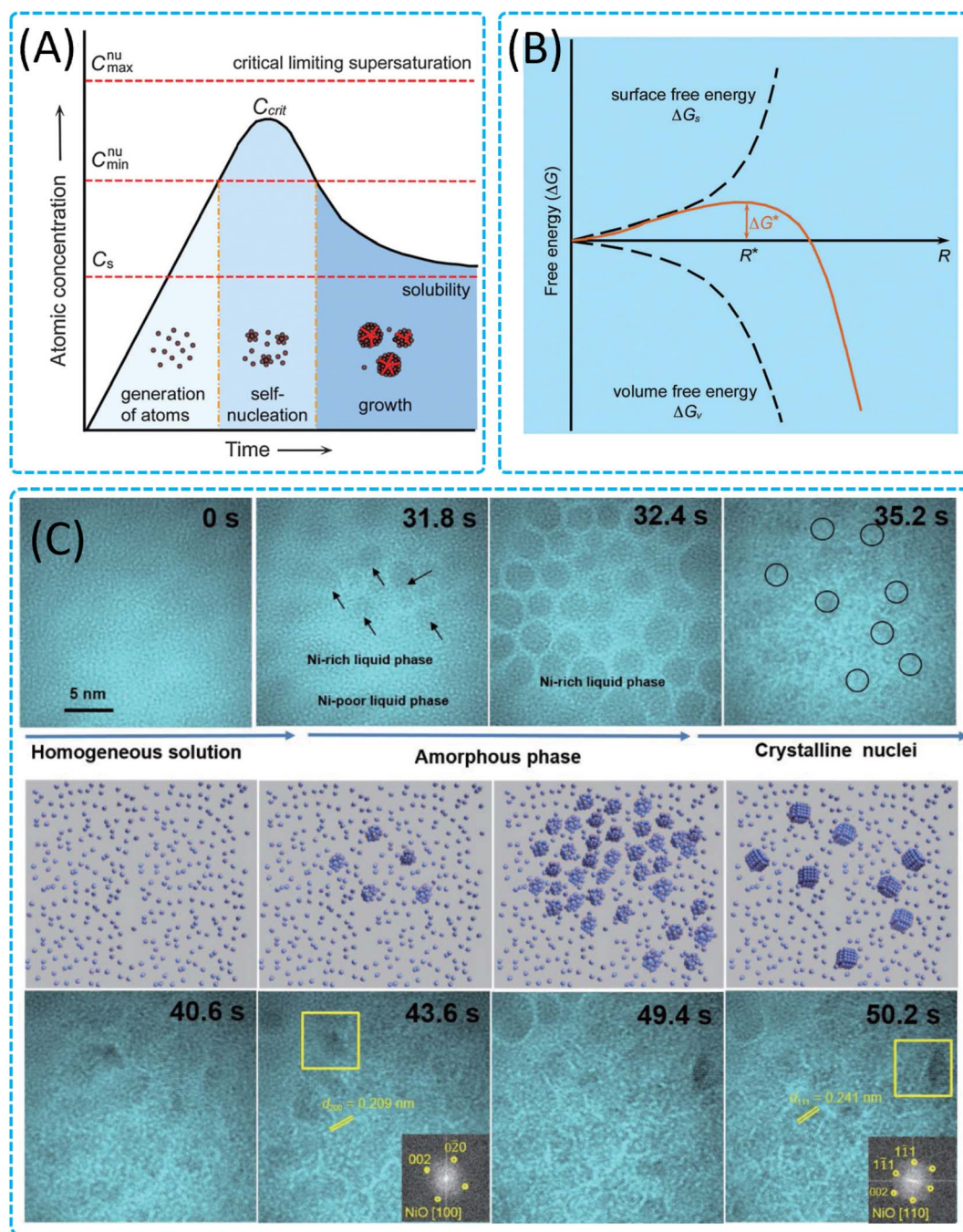
Fig. 3 Nucleation of nanocrystals. (A) LaMer model describing nucleation and growth of nanocrystals as a function of reaction time and concentration of precursor atoms. Reproduced with permission.[58] Copyright 1950, American Chemical Society. (B) Classical nucleation model showing the free energy diagram for nucleation. Reproduced with permission.[59] Copyright 2009, American Chemical Society. (C) TEM images showing the formation of NiO nanocrystals in SiN$_x$ liquid cells. Reproduced with permission.[29] Copyright 2022, Tsinghua University Press.

3.1.2 Thermodynamic driving force. The spontaneous phase transition is driven by the exothermicity of lattice formation. The free energy change (ΔG) for nucleation is determined by the sum of the phase transformation (ΔG_v) and the solid surface formation (ΔG_s). The negative ΔG_v contributes to the system's overall stability, while the positive ΔG_s increases with the surface area of the nuclei (Fig. 3B).[59]

3.1.3 Critical nucleus size. There is a critical size (R^*) at which the total free energy change (ΔG^*) reaches a maximum. Nuclei smaller than this critical size are unstable and tend to dissolve, while larger nuclei are more stable and can grow.

3.1.4 Kinetics of nucleation. The nucleation rate is influenced by the mobility of precursor species and the temperature of the reaction. The Arrhenius equation is used to describe the temperature dependence of the nucleation rate, where an increase in temperature can significantly affect the rate due to the increased mobility of precursors.

3.1.5 Supersaturation. Classical nucleation theory emphasizes the importance of supersaturation for nucleation to occur. A certain concentration of nanocrystals, known as the critical concentration (C_{crit}), must be reached for nucleation to take place.

3.1.6 Energy barrier. There is an energy barrier associated with nucleation, which must be overcome for the process to proceed. This barrier is related to the maximum free energy change (ΔG^*) that occurs at the critical nucleus size.

3.1.7 Growth of nuclei. Once stable nuclei are formed, they can grow at a lower concentration of precursor species that is slightly above the saturation concentration (C_s). This growth process is less energy-consuming compared to the nucleation process.

3.1.8 Control of nucleation. Classical nucleation theory suggests that controlling the nucleation process can be achieved by modulating the surface free energy and/or volume free energy, which in turn affects the total free energy's dependence on the size of the nuclei. This can be done by varying surfactants, forming nuclei of different materials that can transform into the desired nuclei, or changing the reaction environment.

3.1.9 Reversible disorder–order transitions in atomic crystal nucleation. Atomic crystal nucleation, a critical process in materials science, has long been misunderstood. Recent advances in classical nucleation theory have revealed that the nucleation process is more complex than previously thought, involving non-classical mechanisms such as dynamic and reversible structural fluctuations between disordered and crystalline states. *In situ* TEM observations of gold nanocrystal nucleation on graphene surfaces, with millisecond temporal resolution, have shown that the early stages of atomic crystallization are characterized by repeated transformations between these two states.[60] This dynamic process challenges the traditional view of nucleation as a single irreversible transition and highlights the role of metastable states and atomic-scale dynamics in determining the nucleation pathway. The study found that small atomic clusters exhibit a high probability of retaining a disordered state, with the relative population of crystalline states increasing as the cluster grows. Once a certain size is reached, the cluster becomes trapped in the crystalline state due to increased energy differences. This size-dependent thermodynamic stability is attributed to the low energy barrier for crystalline-to-disordered transitions in small nanoclusters, which can be overcome by various energy sources such as monomer attachment and electron beam interactions. These findings not only improve our understanding of the fundamental mechanisms underlying material growth processes such as thin-film deposition, interface-induced precipitation, and nanoparticle formation but also provide insights into the atomic-scale processes that govern material properties.

Classical nucleation theory provides a framework for understanding and controlling the initial stages of nanoparticle formation, which is crucial for the synthesis of colloidal nanoparticles with tailored properties (Fig. 3C).[29]

3.2 Growth of 0D nanomaterials

The realm of zero-dimensional (0D) nanomaterials has witnessed remarkable advancements in both applications and fabrication techniques, marking a significant evolution in nanotechnology.[61] 0D nanomaterials, with their distinct properties arising from quantum confinement effects and large surface-to-volume ratios, have catalyzed a surge of interest across various fields, which encompass energy conversion,[62,63] catalysis,[64,65] photonics,[66] and biology.[67,68] For instance, in the energy sector, these nanomaterials have been utilized in solar cells[69–71] and batteries,[72–74] capitalizing on their size-dependent electronic properties. In catalysis,[75] 0D nanomaterials have demonstrated enhanced activity and selectivity for various chemical reactions, attributed to their well-defined surface structures and compositions. The photonics industry has also benefited from the unique optical properties of 0D nanomaterials, such as in light-emitting diodes and photodetectors.[76,77] Moreover, in the biomedical field,[78] their use in drug delivery systems and imaging agents has shown great promise. The synthesis of 0D nanomaterials has been refined through various approaches, ensuring control over size, shape, and composition–crucial parameters that dictate their properties and performance. Wet chemical methods, including sol–gel, precipitation, and hydrothermal synthesis, have been refined to produce monodisperse 0D nanomaterials with high yields.[15,79,80] Advanced physical techniques such as sputtering and laser ablation have also been employed to synthesize nanomaterials with unique properties.

Furthermore, the integration of *in situ* TEM allows researchers to directly visualize the dynamic processes of nanomaterial synthesis, including nucleation, growth, and phase transformations, which are crucial for understanding the underlying mechanisms and optimizing the material properties.[56,81,82] The ability to apply various external stimuli, such as mechanical, thermal, electrical, and chemical influences, within the TEM environment provides a unique platform for the precise control and tuning of the nanomaterial properties. Advancements in *in situ* TEM techniques have led to the development of new experimental setups that enable the study of 0D nanomaterials under more realistic and complex

conditions. These include the use of nanomanipulation for mechanical testing, MEMS for precise temperature control, and environmental cells for gas and liquid exposure.[23,52,53,83] Furthermore, the integration of spectroscopic tools like EELS and EDS within the TEM has enhanced the chemical and electronic characterization capabilities. Looking forward, the direction of research in *in situ* TEM for 0D nanomaterials is expected to focus on achieving higher spatial and temporal resolutions, improving the stability and controllability of external stimuli, and expanding the range of accessible materials and conditions. The development of novel *in situ* holders and the integration of additional physical fields, such as magnetic and optical stimuli, will further broaden the applicability of *in situ* TEM in the synthesis and study of 0D nanomaterials. Moreover, the combination of *in situ* TEM with computational modeling and simulation will provide deeper insights into atomic-scale processes and enable the prediction and design of new materials with tailored properties.

3.2.1 Growth of 0D nanomaterials through solid-state interactions. In the field of materials science and nanotechnology, a thorough understanding of the control of nanoparticle size and morphology during high-temperature solid-phase synthesis, structural changes, phase transitions and the stability of surface composition and morphology is essential.[84,85] These factors significantly affect the performance and long-term stability of materials in applications such as fuel cell catalysis. In this paper, we comprehensively discuss a number of key factors affecting the structure and performance of materials at high temperatures, including grain growth,[85–88] thermal stress-induced phase transitions,[89,90] agglomeration or sintering of nanoparticles,[47,91] diffusion processes in solid phase reactions,[92–94] compositional inhomogeneities in multi-component systems,[95,96] oxidation and impurity introduction at elevated temperatures,[97,98] and phase transition[99] issues detrimental to late-stage catalytic reactions. The *in situ* heating techniques was used to observe structural changes in nanoparticles, in particular the study of performing *in situ* heating of $PtNi_{1.5}$ octahedral nanoparticles within TEM to study their compositional and morphological changes.[98] These studies provide new insights into the factors influencing catalyst activity and stability and reveal the role of metal clusters in catalysis and crystal nucleation. In addition, Xia *et al.* documented the synthesis of Ru octahedral nanocrystals, thermal stability at elevated temperatures, migration and morphological evolution of metal clusters.[100] Gatalo *et al.* investigated $PtCu_3/C$ and PtM-based alloy nanoparticles as oxygen reduction reaction (ORR) electrocatalysts, providing a significant scientific foundation for the design and optimization of ORR electrocatalysts in fuel cells.[101] Furthermore, the exploration and study of the thermal stability of gold (Au) nanoparticles,[88] the thermal evolution of C–Fe–Bi nanocomposite[92] systems and the dynamic changes of Pt–Co nanoparticles[102,103] during heat treatment provide an important theoretical basis for understanding and controlling the microstructural evolution of materials. The results of these studies are important for the development of new carbon-based materials, optimization of catalyst properties

and understanding the formation mechanism of intermetallic compound nanoparticles.

Meanwhile, studies on the thermal stability and growth dynamics of the structures of novel functional nanoparticles (bimetallic Janus nanostructures (JNs)[32,104–106] and high-entropy oxides (HEOs)[97]) have extended our understanding of the evolutionary behavior of nanoparticles at elevated temperatures and provided important scientific insights into the control of the nanoparticle structure and morphology. These findings not only deepen our understanding of the behavior of nanoparticles at high temperatures, but also provide an important scientific basis for the design of materials with specific properties, especially for applications in catalysts, energy storage, and conversion. JNs have garnered significant attention due to their distinctive interfacial properties that can be finely tuned for a myriad of applications. The size of these nanoparticles plays a pivotal role in dictating the structural and orientational characteristics of their heterointerfaces, which in turn significantly influences their performance in practical applications. Sun *et al.* utilized *in situ* annealing HRTEM and revealed a novel sub-10 nm heterostructure with a unique interface, which offered fresh insights into the role of particle size in interfacial evolution during thermal annealing (Fig. 4A).[32] The study underscores the importance of understanding the atomic motion mechanism that governs the formation of different heterointerfaces, influenced by particle size. The findings are instrumental in the development of nanoelectronic devices and catalytic systems where precise control over the heterointerfaces is essential for optimizing performance. In the realm of materials science, HEOs have emerged as a novel class of multi-component solid-solution materials, showcasing immense potential for applications in catalysis, energy storage, and thermal barrier coatings. The ability to fine-tune their composition and crystal structures offers a vast landscape for enhancing material properties. However, unraveling the atomic-scale mechanisms of nucleation and growth of HEOs has been a formidable challenge, impeding the rational design of their structure and function. Gao *et al.* leveraged atomic resolution *in situ* STEM to visualize the entire formation process of a high-entropy fluorite oxide (HEFO) from a polymeric precursor, as schematically shown in Fig. 4B.[97] The findings of this study are pivotal, providing critical insights into the rational synthesis of HEOs with controlled grain sizes and morphologies, which in turn are essential for tailoring the material's properties. The research demonstrates that the random and uniform distribution of elements in the designed polymeric precursor is fundamental to the low-temperature oxidation and nucleation process. Furthermore, the study elucidates that the formation of HEFO entails slow grain growth through atom diffusion at temperatures below 900 °C and a subsequent liquid-phase-assisted grain growth process at elevated temperatures. This work not only advances the scientific understanding of HEOs but also opens new avenues for the development of advanced materials with targeted properties for specific technological applications.

3.2.2 Growth of 0D nanomaterials in the liquid phase. Liquid-phase preparation of nanoparticles allows precise
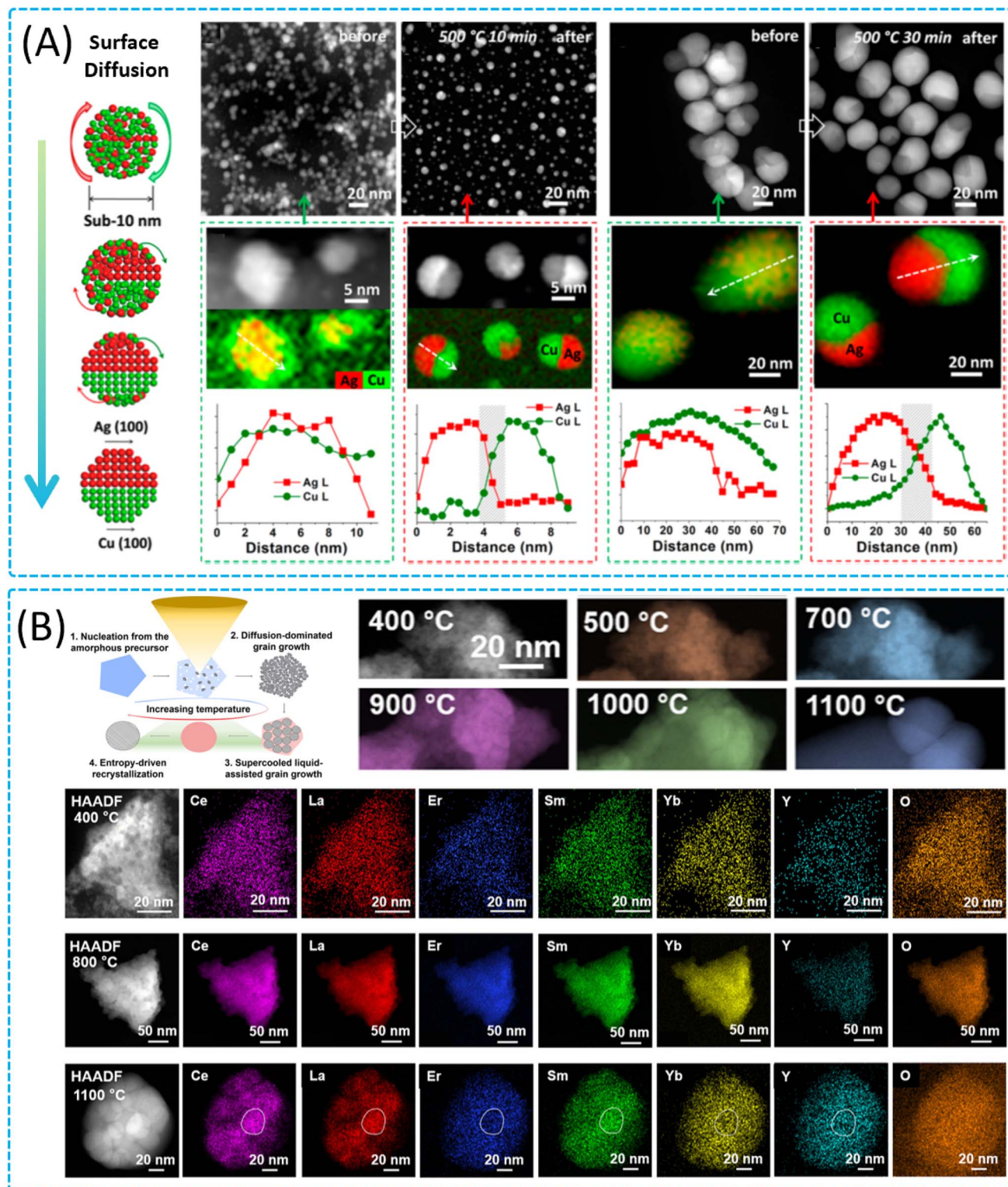
Fig. 4 Phase transition of 0D nanomaterials in solid-state interactions. (A) Schematic representations of the reasonable atomic motion in sub-10 nm CuAg nanoparticles during annealing to form a Cu (100)/Ag (100) heterointerface, and STEM-HAADF images of CuAg nanoparticles before and after annealing under vacuum at 500 °C for 10 min. The corresponding high-magnification STEM-HAADF images and EDS elemental maps. Reproduced with permission.[32] Copyright 2019, American Chemical Society. (B) Schematic representations of the HEFO from an amorphous precursor and characterization of the structures and compositions of high-entropy fluorite oxides at different temperatures. Reproduced with permission.[97] Copyright 2022, American Chemical Society.

regulation of functional materials at the molecular level, which is crucial for the realization of "atom-to-atom" design and construction of materials. The growth mechanism of nanoparticles in the liquid phase is a complex and delicate dynamic process that involves multiple stages, including nucleation,[107] growth,[108] and ultimately the maintenance of particle stability.[29,109] Currently, the growth mechanisms of nanoparticles in the liquid phase mainly include the surface reaction mechanism,[81,110] monomer diffusion mechanism,[111] LaMer theory,[9,58] Ostwald ripening and digestive ripening,[112,113] Finke–Watzky mechanism,[114,115] oriented attachment mechanism,[116–118] intraparticle growth,[119,120] core–shell growth[121–125] and interfacial phase growth.[126] The growth mechanism of nanoparticles in the liquid phase not only determines the size and morphology of nanoparticles, but also affects their physicochemical properties.

As an important class of nanomaterials, twinned noble metal nanoparticles show a wide range of applications and significant technological advantages in many fields, such as catalysis, optics, electronics, and biomedicine, due to their excellent physicochemical properties and tunable surface plasmon resonance.[127–130] The growth process of such noble metal nanoparticles in liquids is mostly kinetically controlled, with a large deviation from the equilibrium state defined by thermodynamics. When the temperature increases, various changes in the arrangement of atoms occur, which affect the geometrical morphology of the nanocrystals, the spatial distribution of the elements, the internal structure and the phase structure. Ma et al. used an in situ LCTEM technique to study in depth multi-twinned gold nanoparticles, especially decahedra and icosahedra, which have a wide range of applications and significant technological advantages (Fig. 5A).[131] The growth of multi-twinned nanoparticles mainly proceeds through two paths: path A involves core-based laminar growth, starting from rounded multi-twinned seeds and growing in a laminar manner (Fig. 5B and C), and path B involves continuous twinning and tetrahedral (tetrahedra) growth (Fig. 5D). There were differences in the internal strain relaxation mechanisms and growth kinetics of the two pathways: in path A, multi-twinned nanoparticles grew by opening and closing of re-entrant grooves at twin boundaries, which was not found in path B. The researchers discussed the preferred pathway (path A) and the preferred pathway (path B) further. The researchers further discussed the relationship between the preferred path (A or B) and the initial seed yield as well as the size- and morphology-dependent multi-twinned nanoparticle formation, revealing the mechanism of formation and evolution of multi-twinned structures. Zhu et al. conducted an in situ study of the growth mechanism of citrate-stabilized gold nanoparticles by oriented attachment (Fig. 5E).[132] It was observed that the process of oriented attachment between pairs of nanoparticles in the liquid phase was tightly controlled by the adsorbed ligand layer, and when two gold nanoparticles approached a certain distance, their ligand layers started to overlap, resulting in the rotation of the pairs of particles shifting from a random mode to an oriented mode until their <111> crystalline surfaces were completely aligned, followed by a rapid contact. The thickness

and interaction energy of the ligand layer in this process are the key factors determining the oriented attachment behavior of nanoparticles. For the core–shell growth mechanism, Tan et al. thoroughly explored the epitaxial growth process of silver (Ag) on gold nanocubes (Au nanocubes) in solution (Fig. 5F).[124] It was found that the formation of the Au–Ag core–shell structure proceeds through two mechanisms: the first mechanism is fusion, where silver nanoparticles are adsorbed onto Au nanocubes (Fig. 5G), and the second mechanism is monolithic attachment, where silver atoms are epitaxially deposited on Au nanocubes (Fig. 5H). Both paths end up with the same Au–Ag core–shell nanostructure. Further analysis reveals that the growth of silver shells is controlled kinetically and thermodynamically. When the surface diffusion rate is faster than the atomic deposition rate, the reaction is driven by thermodynamics, and the silver atoms diffuse and migrate to the sites with the lowest surface free energy, the growth of the shell layer will proceed along the <100> and <110> directions. However, when the atomic deposition rate is higher than the surface diffusion rate, the reaction is controlled by kinetics, with preferential attachment of silver atoms to certain crystalline surfaces, and the shell layer growth pattern may shift to promote the formation of more complex shapes, such as concave cubes or octopods; the discovery of these phenomena is of great significance for the understanding and optimization of the synthesis process of core–shell nanostructures.

3.2.3 Growth of 0D nanomaterials through gas–solid interactions. The complex dynamic behaviour of nanoparticles in gas–solid reactions mainly involves processes such as alloying,[133,134] segregation,[135] phase transition,[136,137] redox reactions,[138–142] lattice reconstruction[140,143] and morphological evolution,[141,144] which together determine the stability, activity and selectivity of the nanoparticles. Kim et al. found that when heated in an oxygen atmosphere, the surface of Pt_3Ni nanoparticles undergoes reconfiguration (Fig. 6A), resulting in the formation of a $Pt–NiO_{1-x}$ interfacial structure.[145] The newly formed $Pt–NiO_{1-x}$ interfacial structure plays a key role in the CO oxidation reaction and promotes the improvement of catalytic activity. In addition, with regard to the formation process and catalytic mechanism of noble metal single atoms, which has received much attention, the structural changes of Pt nanoparticles were observed in real time by ETEM by the Tilley group (Fig. 6B).[146] The thermodynamically driven rearrangement process of Pt islands on low-index Ru crystal surfaces was observed by heating Pt–Ru nanomaterials in an H_2 atmosphere, and the Pt atoms were gradually and uniformly dispersed on the surface of the Ru nanoparticles as the reaction time progressed, eventually forming discrete individual Pt atoms. This change was driven by the combination of decreasing the surface free energy of Pt islands and increasing the strong bonding interaction between Pt and Ru, which contributed to the final dispersion of Pt atoms on the surface of branched Ru nanoparticles and the formation of Pt single atom catalysts with high activity and resistance to CO poisoning. Meanwhile, the phase separation mechanism of the high-entropy alloy nanoparticles in high-temperature environments under different atmosphere bars was thoroughly investigated by the group of Zachariah.
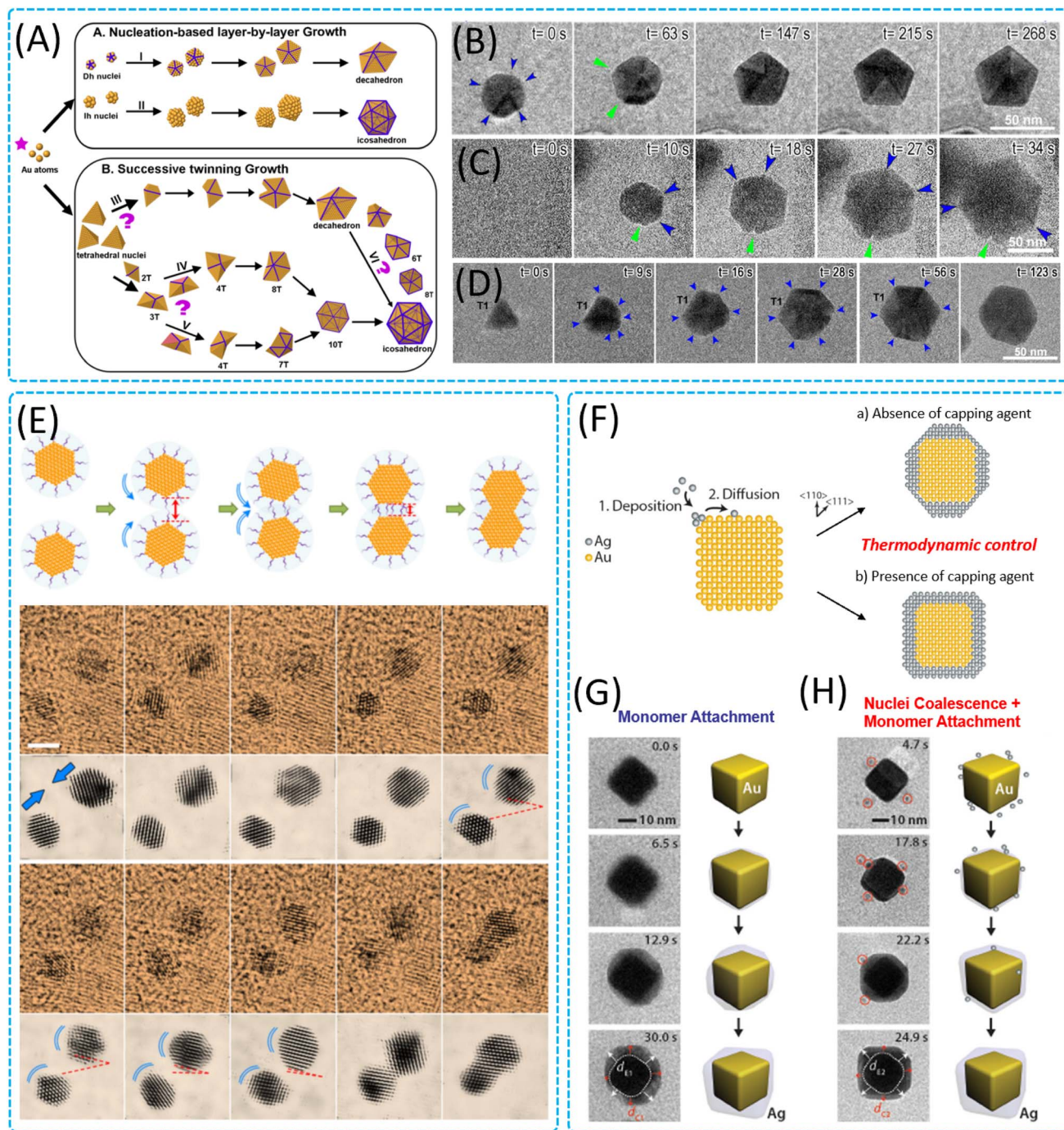
Fig. 5 Nucleation, growth and structural transformation of 0D nanomaterials in the liquid phase. (A) Schematic of different formation routes of decahedral and icosahedral multiply twinned nanoparticles. Time-sequenced TEM images of nucleation-based layer-by-layer growth of decahedral and icosahedral Au NPs from the rounded multiply twinned seeds viewed along (B) [011] and (C) [111] orientations, respectively. (D) Time-sequenced TEM images of successive twinning growth of an icosahedral Au MTP from an initial tetrahedral seed viewed along [111] orientation. Reproduced with permission.[131] Copyright 2020, American Chemical Society. (E) Schematic illumination of the whole oriented attachment process and imaging of oriented attachment at the atomic level. Reproduced with permission.[132] Copyright 2018, Springer Nature. (F) Schematic illustrations showing the shape evolution of a cubic seed under thermodynamic control for two conditions: absence or presence of capping agents. Time-lapse TEM images of Au nanocubes interacting with the Ag precursor solution inside a flow cell via (G) the monomer attachment process and (H) monomer attachment and nuclei coalescence. Reproduced with permission.[124] Copyright 2016, American Chemical Society.

They inspected the high-temperature reduction dynamics of oxidised FeCoNiCuPt high-entropy alloy nanoparticles[147] in a hydrogen environment and found that the outer surface of the oxide layer of the oxidised FeCoNiCuPt high-entropy alloy was transformed into a porous structure in a H_2 atmosphere, where the oxidised state Cu was completely reduced to Cu NPs, while
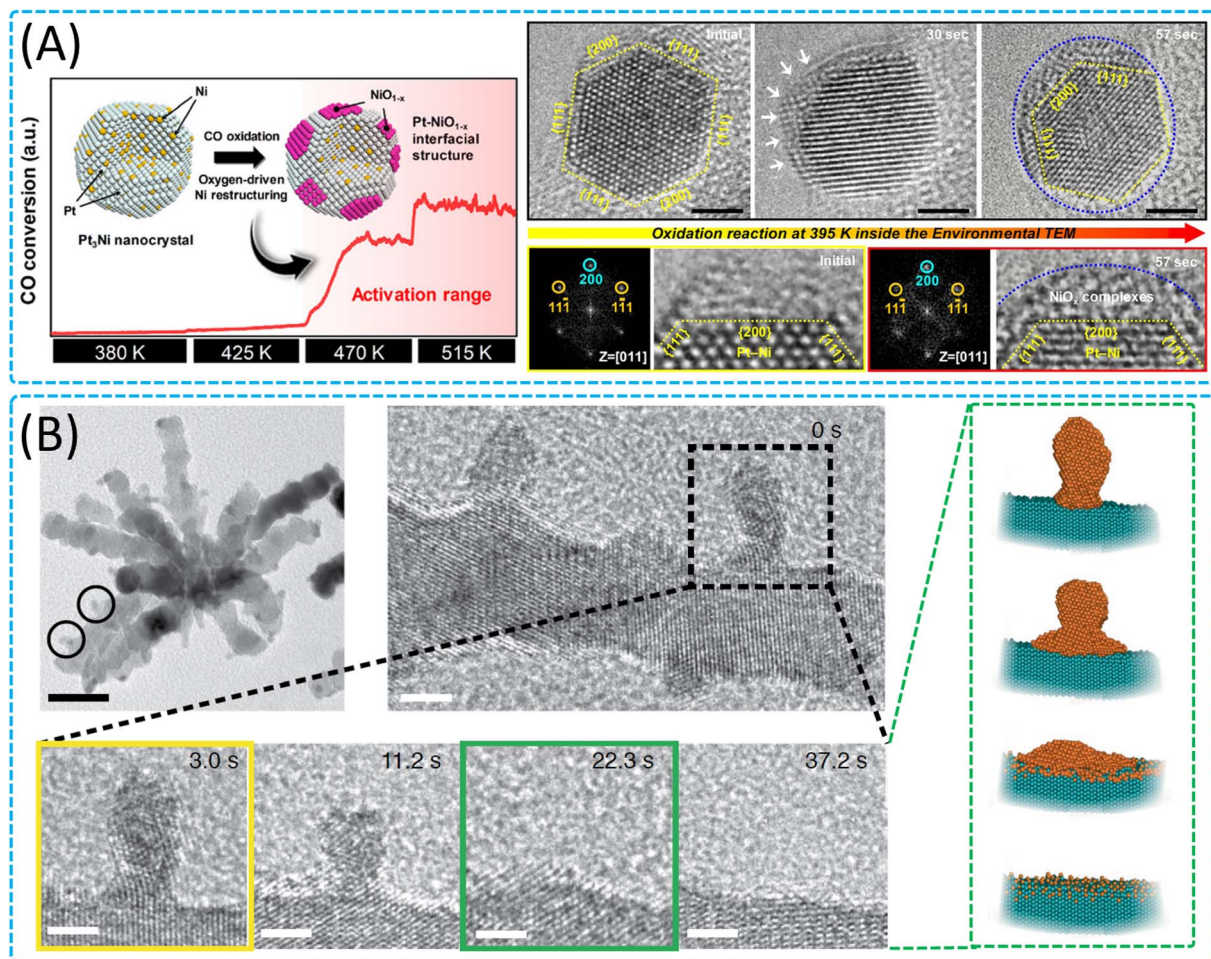
Fig. 6 Structural and phase transformation of 0D nanomaterials in gas−solid interaction. (A) Pt−NiO$_{1-x}$ interfacial structure formation of a Pt$_3$Ni nanoparticle after O$_2$ annealing and *in situ* TEM of the spreading process. Reproduced with permission.[145] Copyright 2020, American Chemical Society. (B) The growing and spreading process of Pt islands on Ru branched nanoparticles to create single-Pt-atom-on-Ru catalysts, and the HRTEM images and corresponding cartoons of the island as it transforms during *in situ* heating under a flow of H$_2$. Scale bars, 2.5 nm. Reproduced with permission.[146] Copyright 2022, Springer Nature.

Fe, Co and Ni remained in the oxidation state. During this process, the core of the oxidized FeCoNiCuPt high entropy alloy nanoparticles shrinks due to the outward diffusion of the transition metals and their associated vacancies, leading to the formation of a gap between the core and the inner surface of the oxide layer, which expands due to the outward diffusion fluxes of all the transition metals (Fe, Co, Ni, and Cu). At the same time, they thoroughly investigated the phase separation mechanism of Fe$_{0.28}$Co$_{0.21}$Ni$_{0.20}$Cu$_{0.08}$Pt$_{0.23}$ high-entropy alloy nanoparticles[148] during high-temperature oxidation and found that their oxidation at 400 °C was controlled by the Kirkendall effect. Transition metals (Fe, Co, Ni and Cu) diffuse outwards to form disordered oxide layers, and locally ordered lattices are observed in the oxides, suggesting that Fe$_2$O$_3$, CoO, NiO and CuO grains are formed in the overall disordered matrix.

3.3 Growth of 1D nanomaterials

1D nanomaterials, with their distinct structural and functional attributes, have emerged as a cornerstone in the realm of nanotechnology.[149] The defining features of 1D nanomaterials include their ultra-high aspect ratios, tunable cross-sectional dimensions ranging from 1 to 100 nanometers, and variable lengths that can span from hundreds of nanometers to millimeters. These attributes endow 1D nanomaterials with a suite of unique properties that are leveraged across a myriad of applications. The high surface-to-volume ratio of 1D nanomaterials facilitates exceptional interaction with their environment, making them ideal candidates for applications in sensing, catalysis, and energy storage. Their quantum confinement effects, stemming from their small cross-sectional dimensions, lead to size-dependent electronic and optical properties, which are pivotal for the development of advanced photonic and electronic devices. Moreover, the crystalline nature of 1D nanomaterials ensures high purity and fewer defects, contributing to their superior performance in various applications. The tunable bandgap and efficient charge transport properties of semiconductor nanowires have been particularly instrumental in the advancement of integrated

photonics, energy conversion, and storage technologies.[150,151] The ability to precisely control the nucleation and growth of these nanomaterials is crucial for tailoring their structures and properties to meet specific application requirements. In this context, *in situ* TEM has emerged as a powerful tool for providing real-time, atomic-scale insights into the formation and evolution of 1D nanomaterials.

In situ TEM allows researchers to directly observe the dynamics of nanocrystal nucleation, growth, and structural transformations under controlled conditions, which is essential for understanding the fundamental mechanisms governing their formation. This technique has been particularly instrumental in studying the effects of various reaction parameters, such as temperature, atmosphere, and substrate interactions, on the nucleation and growth behaviors of 1D nanomaterials. By employing *in situ* TEM, researchers have been able to reveal the atomic migration dynamics, interfacial evolution, and phase transformations during the synthesis of nanowires, nanotubes, and other 1D structures. Advancements in *in situ* TEM have also led to the development of new synthetic strategies, such as catalyst-assisted growth mechanisms, including vapor–liquid–solid (VLS)[152–154] and vapor–solid–solid (VSS)[155–157] pathways, which have been extensively studied for the controlled fabrication of 1D nanomaterials. These *in situ* studies have provided valuable insights into the role of catalysts, the influence of gas environments, and the mass transfer processes during nanowire growth, enabling the optimization of synthesis conditions for the production of high-quality 1D nanomaterials with desired properties. Moreover, *in situ* TEM has been pivotal in exploring the growth of 1D nanomaterials under various conditions, including the effects of strain, defects, and surface/interface engineering. The technique has also been used to investigate the dynamics of light elemental atoms in metal nanocrystals during catalytic reactions, which is crucial for understanding their catalytic performance and potential applications. Looking forward, the continued development of *in situ* TEM techniques, such as the integration of more complex growth environments and advanced characterization methods, holds great promise for gaining a deeper comprehension of the nucleation and growth processes of 1D nanomaterials.

3.3.1 Growth of 1D nanomaterials through solid-state interactions. In the current rapid development of nanoscience and technology, nanowires, which are a type of one-dimensional nanostructure possessing distinctive electrical, optical, and chemical properties, have emerged as a focal point for research and application.[149,150] They exhibit considerable potential in areas such as microelectronics, sensors, optoelectronic devices, and so on.[151,158,159] However, precise control of the growth direction, structure and size of nanowires at the atomic level is very complex and requires an in-depth understanding of interatomic interactions and dynamical processes. Especially in solid-state non-catalytic reaction processes, the origin and migration mechanisms of atoms may be different from catalytic growth, and more theoretical studies are needed to reveal these mechanisms. Using *in situ* heating TEM, researchers have carried out detailed studies on different types

of nanowire growth mechanisms, covering a wide range of aspects such as the growth of metal oxide nanowires,[160] the growth of non-catalytic ZnO nanopillars[161] and Ag_2S-catalysed ZnS nanowires,[162] the in-plane growth of silicon nanowires (Si NWs) with indium (In) as a catalyst,[163] and the evolution of the anisotropy of InAs nanowires[164] and Au nanowires.[48] These studies not only provide an in-depth understanding of the kinetics and physicochemical processes of nanowire growth, but also directly observe the dynamic processes of nanowire growth by *in situ* TEM, revealing key mechanisms such as the microcrucible mechanism, the gas–liquid–solid (VLS) mechanism, and the formation of heterostructure nanowires. Boston *et al.* not only directly observed the nanowire growth process, but also revealed a novel microcrucible mechanism, which is a unique way to promote the growth of metal oxide nanowires at high temperatures, opening up a new chapter in the growth of metal oxide nanowires (Fig. 7A and B).[165] Through a precise experimental design, the team tracked the growth of tetrameric metal oxide nanowires (Y_2BaCuO_5) and discovered the melting and diffusion behaviours of barium carbonate ($BaCO_3$) nanoparticles during heating, which form droplets on the surface of the porous substrate that act as catalytic points, triggering the growth of Y_2BaCuO_5 nanowires. The key features of this microcrucible mechanism include the melting and diffusion of nanoparticles, formation of droplets as a microcrucible and catalytic points, growth of nanowires, maintenance of a fixed diameter growth, dynamics of the liquid–solid interface, and diversity of nanowire morphology. These steps work together to make the nanowire growth process a dynamic and complex process involving interactions between the droplet and the solid matrix. The understanding and application of the microcrucible mechanism provide new strategies for controlling the growth conditions, morphology, and properties of complex oxide nanowires. Chang *et al.* controlled the crystallinity and microstructure of TiO_2/In_2O_3 nanowires through solid-state reactions and observed the In_2O_3/TiO_2 heterostructured nanowire formation process in a high vacuum environment with *in situ* TEM (Fig. 7C).[166] It was found that at the beginning of the reaction, Ti atoms diffused into the In_2O_3 nanowires along the [100] direction to form amorphous TiO_x, and with the annealing process, TiO_x gradually transformed into polycrystalline TiO_2, indicating that the secondary annealing could improve the crystallinity of the material, and the study revealed the effect of different annealing temperatures on the formation of heterostructures of the nanowires and the photoresponse properties of these heterostructures in the enhancement. These *in situ* observations on the growth mechanism of nanowires not only advance the theoretical understanding of nanowire growth, but also provide the scientific basis and technical support for future applications of nanowires in electronics, photonics and other high-tech fields.

3.3.2 Growth of 1D nanomaterials in the liquid phase. The liquid environment provides a mild and controllable growth medium, enabling the growth of nanowires at lower temperatures, and the structure, size, shape, and composition of the nanowires can be precisely regulated by adjusting the conditions of solution components, concentration, reaction
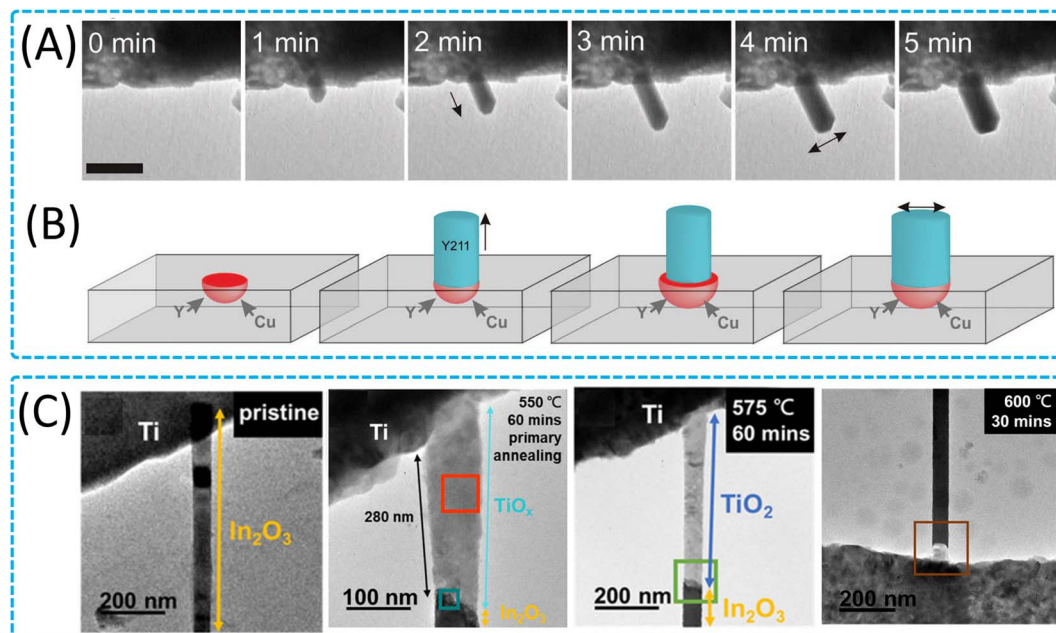
Fig. 7 Growth of 1D nanomaterials through solid-state interactions. (A) TEM images and (B) schematic of a nanowire spontaneously broadening as a result of microcrucible creep and expansion on continuing heating at 500 °C. Reproduced with permission.[165] Copyright 2014, AAAS. (C) Characterization of the pristine In_2O_3 nanowire and different temperature annealed In_2O_3/TiO_x heterostructure nanowire. Reproduced with permission.[166] Copyright 2022, Elsevier.

temperature, and time.[167–169] In liquid-phase reactions, the study of nanowire growth mechanisms helps to deeply understand the physicochemical processes at the nanoscale, explore new synthesis methods and growth mechanisms, and achieve structural control and performance optimisation of nanowires.[170–173] For example, studies on the growth mechanism of catalyst-free nanowires have rarely been reported, so the growth kinetics of catalyst-free nanowires are still unknown.[174] Asghar et al. revealed the microscopic growth mechanism of ceria nanowires through the study of in situ TEM.[175] Meanwhile, Liao et al. revealed the catalyst-free growth of Pt_3Fe nanorods with a noble metal alloy-like material in solution (Fig. 8A).[176] It was found that the growth of nanorods went through three distinct stages: initial nucleation, formation of nanoparticle chains and shaping of nanorods. The Pt and Fe precursors were reduced by electron beam irradiation to form many small nanoparticles, which grew either by monolithic attachment or by merging. The nanoparticles interacted with each other through shape-directed nanoparticle attachment to form nanoparticle chains. The chains were initially curved and polycrystalline but as growth proceeds, they straighten through structural relaxation and eventually form single-crystal nanorods. It was observed that individual nanoparticles or short chains can serve as the basic building blocks for the formation of Pt_3Fe nanowires. During the formation of nanoparticle chains, the relative positions and orientations between nanoparticles change over time, suggesting that the nanoparticles in the chains can undergo relative motion and orientation changes. During the final growth phase, neighbouring nanoparticles in the chain come into contact and form a neck, which

is subsequently eliminated by mass redistribution to form a smooth nanowire. This shows that the complex growth process of constructing one-dimensional nanostructures from nanoparticles mainly involves the key steps of shape-oriented nanoparticle attachment, straightening, orientation correction, and mass redistribution, which provides an important mechanistic understanding for designing hierarchical nanomaterials and controlling nanocrystal self-assembly. In addition, heterostructured nanowires, as an important class of multifunctional nanomaterials, have attracted much attention for their growth process in solution. Niu et al. directly observed and tracked the growth trajectory of lead sulfide on gold nanorod seeds in liquid[177] and found unique metal–semiconductor interface and heterostructure growth dynamics, including volume contraction of the core particles and interfacial strain-driven mass transport and new phase formation, which will help the future design and control of specific heterogeneous nanostructures.[177] The development of in situ TEM has helped to explore new mechanisms of nanowire growth in solution. Cheek et al.[55] discovered a novel electrochemical liquid–liquid–solid growth mechanism for Ge nanowires, where liquid metal nanodrops (gallium or indium) spontaneously react with dissolved GeO_2 in the absence of an external power source, using secondary electrons from electron beam scattering to reduce GeO_2 and facilitate Ge nanowire formation and growth on the liquid metal droplet surface. The growth process is affected by a number of factors, including the surface conditions of the liquid metal nanodrops, the size and density of the liquid metal nanodrops, the concentration of dissolved GeO_2, and the intensity of the electron beam. It was
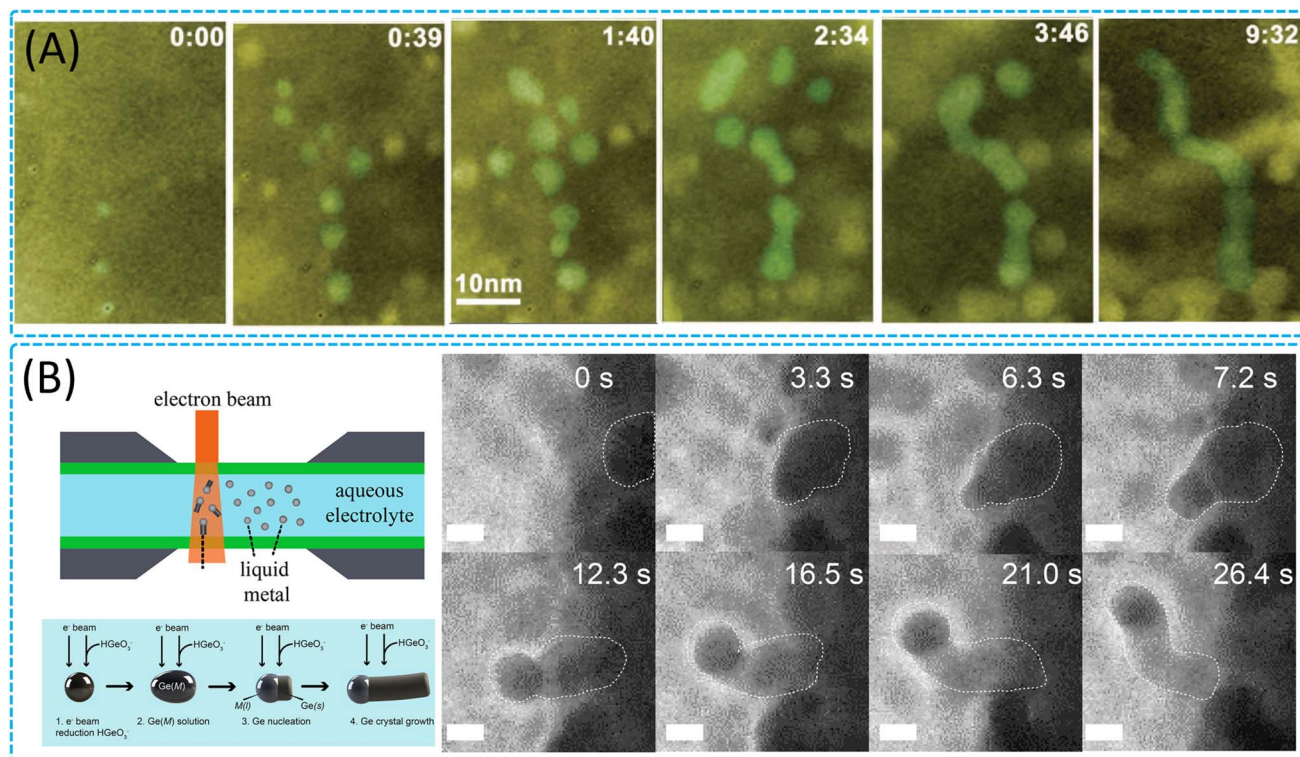
Fig. 8 Growth of 1D nanomaterials in the liquid phase. (A) Sequential color TEM images showing the evolution from the initial nucleation and growth in the molecular precursor solution to a later stage of nanowire formation by shape-directed nanoparticle attachment. Reproduced with permission.[176] Copyright 2012, AAAS. (B) Schematic depiction of Ge nanowire growth and frame grabs with a Ga nanodroplet immersed in an aqueous solution. Reproduced with permission.[55] Copyright 2020, American Chemical Society.

observed in the study that the surface conditions of the liquid metal nanodrops were critical for the growth of the nanowires, and that proper ligand coverage was necessary for growth to occur, whereas excessive coverage of the surface ligands inhibited growth. In addition, the growth rate of Ge nanowires is limited by the rate of Ge supply to the crystal growth front rather than by the rate of crystallization at the liquid metal/solid Ge interface (Fig. 8B), suggesting that the growth process may be controlled by the rate of Ge supply in solution. The nanowire growth conditions away from thermodynamic equilibrium observed in the study provide a new way to control the morphology and crystallographic quality of the nanowires, suggesting that precise tuning of the nanowire structure can be achieved by regulating the growth conditions.

3.3.3 Growth of 1D nanomaterials through gas–solid interactions. The gas–solid reaction growth mechanism of 1D nanowires (gas–solid (VS),[178,179] gas–solid–solid (VSS),[156,180] catalyst-assisted gas–solid growth[180–183] and autocatalytic growth mechanism[179,184]) involves the interaction of a precursor in the gas phase with a solid substrate or a catalyst at high temperatures, which ultimately forms nanowires (metal nanowires,[179,182,184] semiconductor nanowires,[185] carbon nanotubes (CNTs),[30,32,186–191] etc.). These mechanisms not only determine the growth rate, growth direction, diameter, length and crystal structure of nanowires, but also have a profound impact on the properties and applications of nanowires. For the study of the

growth mechanism of metallic nanowires, researchers explored the growth of gold nanowires and platinum-based metal nanowires, where gold nanowires were synthesized by ultrasound-assisted synthesis on plasma-activated graphene templates, and platinum-based metal nanowires were prepared by hydrogen-assisted gas-phase synthesis without the presence of a catalyst. Ma *et al.* found that the growth of platinum-based metal nanowires belonged to the diffusion-assisted solid-state oriented attachment autocatalytic solid-state oriented adhesion of platinum-based metal nanowires.[179] The growth of Pt-based metal nanowires was found to follow a diffusion-assisted autocatalytic growth mechanism of solid-state oriented attachment. This mechanism consists of four stages: the Pt precursors are reduced in a hydrogen atmosphere to form a large number of Pt nuclei, the Pt nuclei gradually grow to form Pt nanoparticles of a few nanometers by consuming smaller nuclei, the Pt nanoparticles are attached and fused to form Pt chains through specific orientation relationships, and these chains are further grown into ultrathin Pt nanowires. During the whole growth process, hydrogen plays a crucial role, which not only promotes the reduction of Pt precursors, but also changes the atomic diffusion rate on the Pt surface, especially on the {100} crystal surface, making the surface diffusion of Pt atoms more significant, thus promoting the formation of nanowires. In the absence of hydrogen, the Pt precursors only aggregated and failed to form nanowires. In addition, the

researchers successfully synthesized Pt–Ni alloy nanowires with excellent electrocatalytic activity and stability, further demonstrating the universality and practicality of this growth mechanism and providing new insights into the structure-controlled synthesis of future metal nanowires. As an important class of one-dimensional nanomaterials, the growth mechanisms (apical growth mode and bottom growth mode) of CNTs directly affect the structural control, quality, and yield of CNTs, which in turn determines their potential applications in various fields. The researchers explored the dynamic behavior of different catalysts during the nucleation of carbon nanotubes by ETEM, including the diffusion of carbon atoms on the catalyst surface and the nucleation at the catalyst–CNT interface. It is shown that the growth of carbon nanotubes involves the diffusion of carbon atoms on the catalyst surface and the structural evolution. Among them, the growth mode of CNTs on catalysts is mainly the tip growth mode, in which the carbon source forms carbon atoms or carbon structures on the top of catalyst particles and gradually extends to form nanotubes. During the growth process, the catalyst particle serves as a template to promote the polymerization of carbon and the extension of the tube (Fig. 9A and B). Yoshida et al.[30] found that for multi-walled carbon nanotubes (MWNTs), a graphene layer first forms on one face of the catalyst (Fe) nanoparticle, and then it gradually extends and bends along the face of the nanoparticle, and a new graphene layer nucleates underneath the existing layer (Fig. 9A), resulting in the nanoparticle's gradual deformation, eventually forming a protrusion from which carbon nanotubes grow rapidly. Moreover, Zhao et al. investigated that the physical state of the catalyst (solid or liquid) has a decisive influence on the diameter and chirality of single-walled carbon nanotubes during the growth process. The diameter of single-walled carbon nanotubes grown on solid catalysts (Co_7W_6) is smaller than that of the catalyst particles, whereas the diameter of single-walled carbon nanotubes grown on liquid catalysts (Co) is similar to that of the catalyst particles (Fig. 9B).[186] Wang et al. explored another growth mode of carbon nanotubes (CNTs),[31] the bottom-end growth mode, and investigated in detail the active phase of cobalt (Co) catalysts and the catalyst-assisted gas–solid growth mechanism during the growth of carbon nanotubes (Fig. 9C). The cobalt catalyst precursor was first deposited on a silicon nitride (SiN_x) film, and metallic Co nanoparticles were formed by oxidation and reduction processes. During CNT growth, the active phase of the Co catalyst was the orthorhombic phase Co_3C, and carbon atoms were supplied by diffusion through the surface and the nanoparticles–CNT interface rather than through the bulk phase of Co_3C. They observed that the growth of CNTs followed a bottom-end growth pattern, where the growth of carbon nanotubes was guided by the bottom end connected to the catalyst particles. During the nucleation phase, the metal fcc Co nanoparticles rapidly transformed into orthorhombic phase Co_3C nanoparticles, followed by the gradual formation of a graphite layer on the surface of the Co_3C nanoparticles until the CNTs began to grow. The growth of the outer wall of MWCNTs is supplied by surface-diffused carbon atoms, while the growth of the inner wall may be realized by interfacial diffusion between the catalyst and CNTs. Thus, in situ TEM plays a key role in understanding the multiple growth mechanisms of metal-based, semiconductor-based and carbon-based nanowires and provides new ideas and insights for the structure-controlled synthesis of future nanowires.

3.4 Growth of 2D nanomaterials

2D nanomaterials are characterized by their ultrathin profiles and exhibit distinctive properties due to quantum confinement effects, which differentiate them from their bulk forms.[192–194] These materials demonstrate remarkable potential in various high-tech applications, including next-generation electronics, optoelectronics, magnetism, spintronics, catalysis, and energy storage due to their unique geometric structures and extraordinary properties.[195–197] The functionality of 2D nanomaterials is strongly related to their atomic structure, morphology, and the presence of defects and interfaces, which can be precisely engineered. Defects, which are common in both natural and synthesized crystals, can act as active sites, introducing localized electronic states and significantly enhancing the properties of the 2D materials. Interfaces between 2D materials and other components, as well as heterostructures created by stacking different 2D materials, can exhibit unique electronic and optoelectronic properties.[198]

In situ TEM has emerged as a cutting-edge technique for studying the growth mechanisms of 2D nanomaterials at the atomic scale.[199,200] It provides real-time monitoring capabilities under various stimuli, including electron irradiation, thermal excitation, and voltage bias, offering unprecedented insights into nucleation, growth, and phase transformations.[19,201] Advanced techniques such as aberration-corrected STEM and EELS have further enhanced the understanding of the structure–property relationships in 2D materials by enabling detailed examinations of their atomic structures, chemical compositions, and electronic properties.[46,202] The current state of research leveraging in situ TEM showcases the ability to identify various polymorphs, defects, and interfaces in 2D nanomaterials. It also highlights the capacity for atom-by-atom chemical analysis and the study of excitons and phonons, which are crucial for understanding the properties of the 2D nanomaterials.

3.4.1 Growth of 2D nanomaterials through solid-state interactions. In the field of exploring the synthesis of 2D materials, the high-temperature thermal decomposition method has garnered widespread attention due to its simplicity, scalability, and potential for precise control over growth conditions. This method utilizes the decomposition reaction of solid precursors under high-temperature conditions to generate 2D materials with specific structures and properties.[200,203] However, variations in the interfacial migration rates[204] and the presence of defects[205] in the 2D materials during the high-temperature thermal decomposition process may affect the quality and performance of the materials. At the same time, although solid-phase high-temperature thermal decomposition reactions can grow 2D materials, understanding of the dynamic evolution and mechanisms during growth is still limited, which
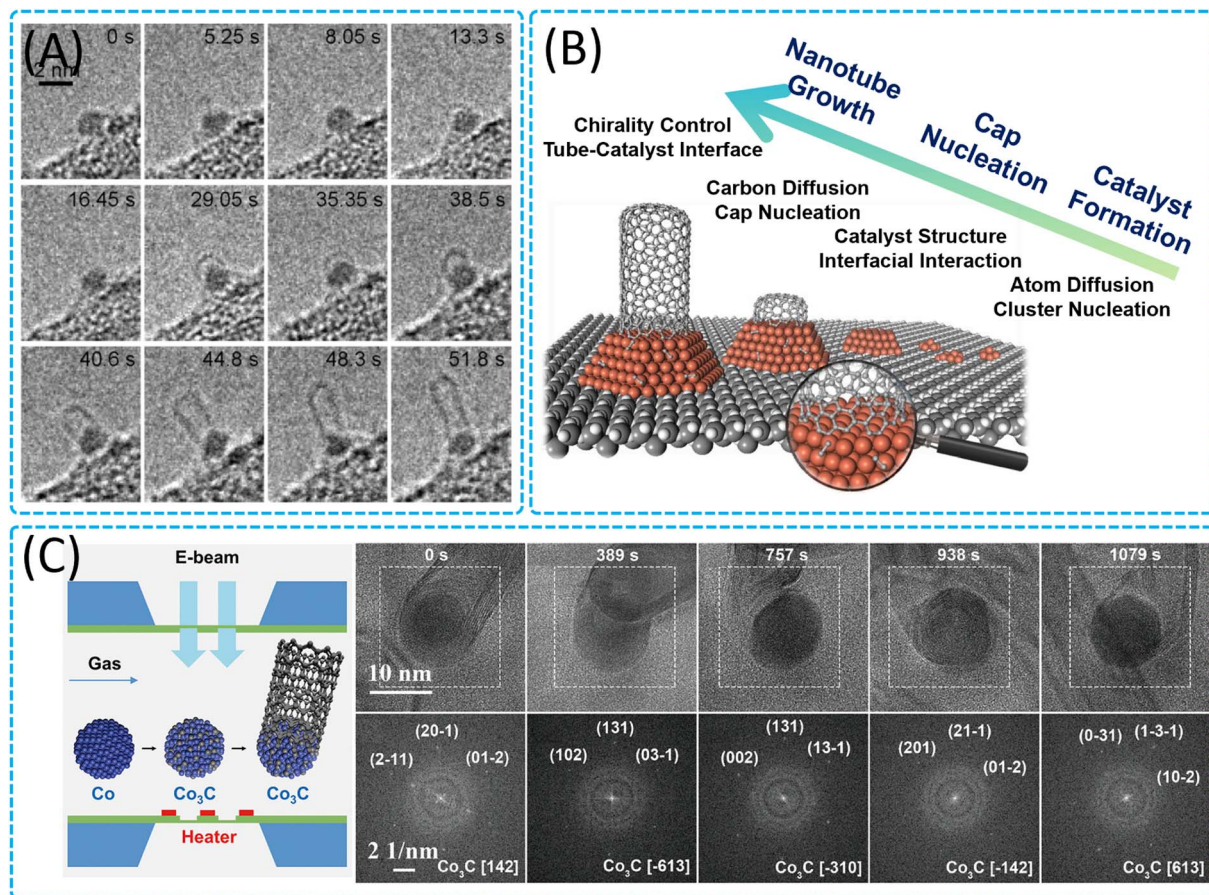
Fig. 9 Catalyst-assisted 1D nanomaterial growth through gas–solid interactions. (A) Nucleation and growth process of a single-walled carbon nanotubes from a nanoparticle catalyst on a substrate. Reproduced with permission.[30] Copyright 2008, American Chemical Society. (B) Schematic illustration of the catalyst evolution and growth process of single-walled carbon nanotubes as revealed by *in situ* ETEM. Reproduced with permission.[186] Copyright 2022, American Chemical Society. (C) The phase evolution of cobalt catalyst nanoparticles during the incubation, nucleation, and growth stages of carbon nanotubes under near-atmospheric pressure using an *in situ* close-cell ETEM and phase structure of an active cobalt catalyst nanoparticle during carbon nanotube growth. Reproduced with permission.[31] Copyright 2020, American Chemical Society.

restricts precise control over material growth. The application of *in situ* heating sample holder technology in TEM provides a new perspective for the preparation and characterization study of 2D materials. It allows us to directly observe the growth process of 2D materials at the atomic scale, including nucleation, growth, and phase transitions of crystals, which is crucial for understanding the formation mechanisms of the materials. Additionally, by precisely controlling conditions such as heating rates and temperatures, it is possible to study the synthesis process of materials under simulated actual growth conditions, thereby achieving precise control over the structure and properties of the materials.

Researchers have utilized *in situ* TEM with an *in situ* heating holder to delve into the growth mechanisms of various 2D nanomaterials (WS$_2$,[206] MoS$_2$,[49,51,203,204,207–209] V$_2$O$_5$ (ref. 50), *etc.*) at high temperatures during solid-state reactions. The study reveals precise control over the vertical and horizontal growth of different 2D materials through the thermolysis of solid precursors, as well as multiple growth stages and various growth modes formed on different substrates. Gavhane *et al.* achieved control over the vertical and horizontal growth of WS$_2$

by altering the thickness of the precursor (Fig. 10A–C),[206] and the study found that on different metal-deposited heating chips, two layers of WS$_2$ formed interference patterns by rotating at various angles relative to each other, providing a new perspective for understanding the growth dynamics of WS$_2$. Additionally, Kondekar *et al.* discovered that a low concentration of Ni can significantly alter the crystallization and growth process of MoS$_2$, leading to an increase in MoS$_2$ crystal size, which may be due to changes in the migration rate of grain boundaries during the growth process (Fig. 10D).[203] In contrast, a higher concentration of Ni inhibits the formation of MoS$_2$, instead forming Ni and nickel sulfides. These findings indicate that the addition of other metal elements during synthesis plays a crucial role in the evolution of 2D nanomaterials. Regarding 2D V$_2$O$_5$ nanomaterials, Gavhane *et al.* utilized an *in situ* heating holder to observe in real time the formation process of two-dimensional V$_2$O$_5$ nanostructures, including the growth of orthogonal V$_2$O$_5$ 2D nanosheets and 1D nanobelts.[50] The study also revealed the phase transition process of V$_2$O$_5$ to VO$_2$ and optimized the temperature range required for the growth of V$_2$O$_5$ nanostructures. These studies provide in-depth insights
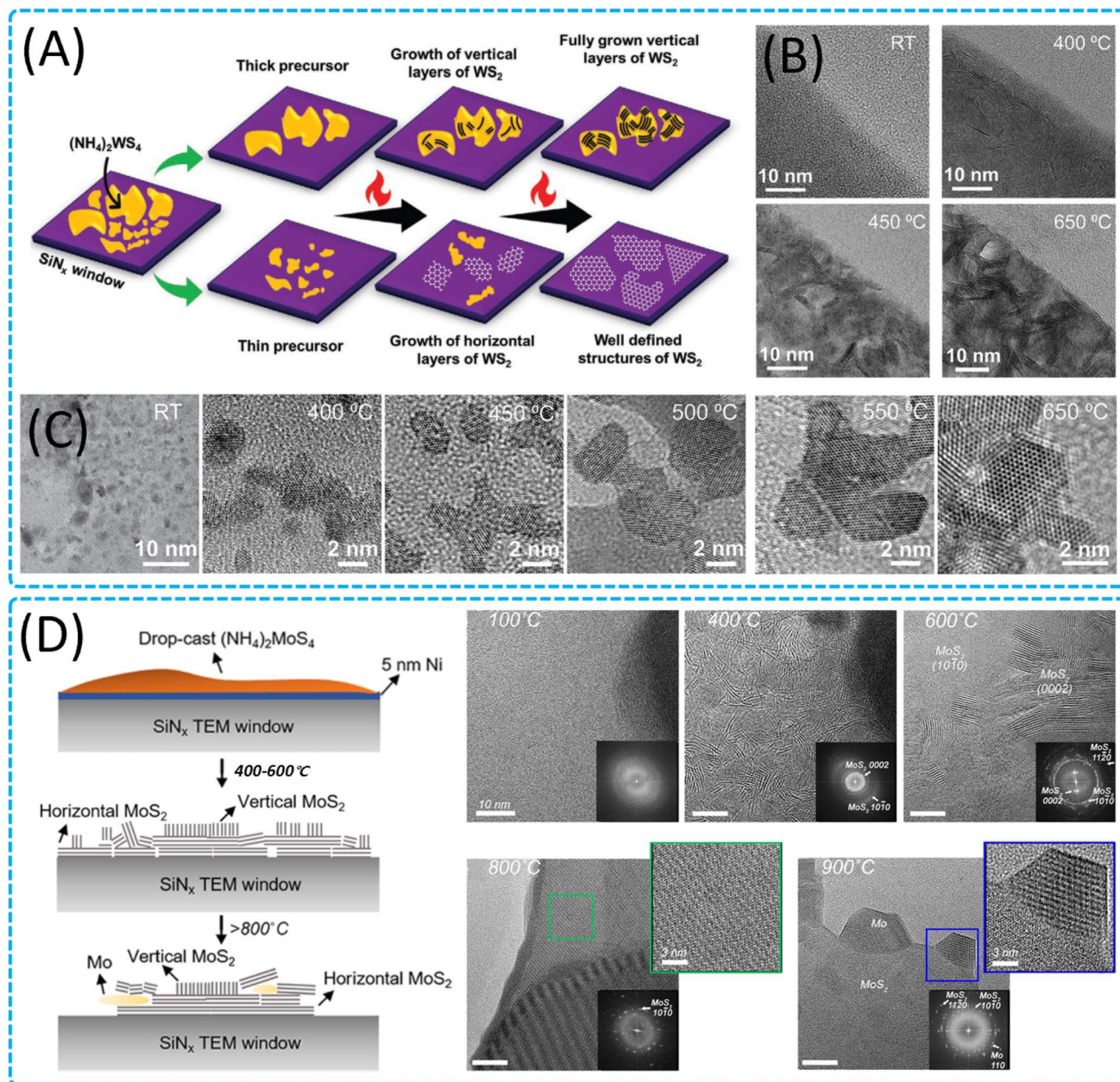
Fig. 10 Structure evolution and growth of 2D nanomaterials through solid-state interactions. (A) Schematic illustration of the evolution of polycrystalline WS_2 through thermolysis of an ammonium tetra-thiotungstate precursor, showing the growth of vertical layers and horizontal structures with heating in thick and thin precursor areas, respectively. (B) Growth of vertically aligned layers of WS_2 at different temperatures. (C) Horizontal growth of WS_2 layers at different temperatures.[206] Copyright 2022, Wiley. (D) Illustration summarizing the formation of MoS_2 crystals from the pure ammonium tetrathiomolybdate precursor and *in situ* TEM images showing the evolution of the ammonium tetrathiomolybdate precursor in the presence of a 5 nm Ni film during heating to different temperatures.[203] Copyright 2019, American Chemical Society.

into the growth dynamics of WS_2, MoS_2, and V_2O_5 and offer effective pathways for the preparation of 2D nanomaterials. Furthermore, Kotakoski *et al.* employed *in situ* STEM combined with a deep learning framework to explore the dynamic process of MoS_2 restructuring from 2D to 3D configurations and its growth on graphene.[209] These studies not only provide an in-depth atomic-level understanding of the growth and structure of 2D nanomaterials but also demonstrate the potential of deep learning technology in 2D material research, offering new

avenues for exploring novel structures and properties. Concurrently, the study investigated the dynamic behavior and structural changes of nanocrystalline graphene under high-temperature conditions, as well as the structural evolution of vertically aligned 2D MoS_2 layers, providing key insights into the structural stability of general van der Waals 2D crystals and offering valuable technical guidance for material design and optimization. Overall, the results from *in situ* heating studies on pure solid-state reactions are crucial for guiding the synthesis,

manufacturing, and customization of functional characteristics of 2D nanomaterials, providing new understanding for the controlled synthesis of large-area 2D nanomaterials, and holding the potential for achieving atomic-level precise control and growth of 2D nanomaterials during solid-state reactions.

3.4.2 Growth of 2D nanomaterials in the liquid phase. Liquid-phase preparation methods allow precise control of the size, morphology and structure of 2D nanomaterials at the molecular level, enabling fine-tuning of the electronic, optical and catalytic properties of 2D materials.[210] The development of *in situ* liquid-cell TEM technology has enabled real-time imaging of the reaction process in a liquid environment with high spatial and temporal resolution, providing a new perspective on molecular-scale dynamics and a deeper understanding of the growth mechanisms and morphological evolution of materials, thus optimising synthesis conditions and improving material quality and yield. These *in situ* studies mainly include the growth mechanisms of 2D nanomaterials (metal monolithic 2D materials,[211–213] 2D transition metal oxides[214–216] and sulfur compounds,[215] core–shell 2D materials,[217] core–shell structures,[218,219] MOF-based 2D structures.[220–223] *etc.*), the kinetic pathways during crystal formation, and the superstructure assembly of 2D materials.[223,224] The formation mechanism of conventional 2D materials such as graphene, metal hydroxides[225] or sulfur compounds is relatively well defined due to their inherent layered atomic structure. However, the mechanism of how non-lamellar crystals form 2D nanosheets is not clear. Zheng's group thoroughly investigated the formation process of non-lamellar transition metal oxide nanosheets and revealed the growth mechanism of how non-lamellar crystals form 2D nanosheets (Fig. 11A).[215] Experimentally, it was observed that under certain conditions, 3D nanoparticles start to grow as the reaction time progresses. During the transition, specific crystal planes grow at different rates, resulting in nanoparticles with specific morphologies, such as 'butterfly' shapes or squares. Competition between surface and edge energies, size effects, and ligand–surface interactions are the main factors leading to the transition from 3D nanoparticles to 2D nanosheets.

The structure–property relationship of noble metal nanocrystals is crucial for their applications in various fields such as catalysis and sensing. Noble metal nanosheets exhibit unique behaviours during shape transformation due to their high surface-to-volume ratio and dynamic surface reactivity, which include adsorption, desorption, and diffusion of surface atoms, processes that are critical for the overall shape change. Therefore, researchers have delved into the growth kinetics and formation mechanisms of gold nanosheets,[212,213,227] silver nanosheets,[228] and palladium dendritic nanosheets.[211] Alloyeau *et al.* found that the growth of colloidal nanoparticles is affected by a combination of kinetic and thermodynamic effects and that by controlling the electron dosage, it is possible to control the growth rate directly, thus quantifying the influence of kinetic effects on planar nanoparticle formation.[227] Park *et al.* revealed that at lower electron doses, the growth of gold nanosheets is driven by thermodynamics, and the formation and shape of nanosheets are directly related to the formation of

twinned surfaces during growth.[212] Jin *et al.* also found that the growth rate of Au nanocrystals can be precisely controlled by adjusting the solution chemistry, in particular pH and chloride ion concentration, which is important for the design and synthesis of nanostructures with specific shapes and structures.[213] In addition, Choi *et al.*[226] explored the shape change mechanism of Au nanosheets (Fig. 11B–E) and found that the diffusion of nanocrystal surface atoms is the main determinant of the final structure in the shape change and that this rapid diffusion of surface atoms leads to a truncated morphology transition of unstable crystal surfaces, thus minimising the surface energy. Liquid *in situ* experiments revealed that oxidative etching of gold nanoprisms and subsequent structural remodelling of the crystal faces were induced by changing the chemical potential in the reaction solution and that diffusion of surface atoms on the exposed crystal faces led to the development of unstable {220} crystal faces into stable {111} crystal faces, resulting in truncated morphologies with minimal surface energy. This finding not only provides a new perspective for understanding the formation mechanism of nanocrystals of various shapes, but also has important implications for the controlled synthesis of colloidal nanocrystals. Meanwhile, for silver nanosheets, E studies revealed a dissolution-re-growth mechanism from triangular to hexagonal shapes, providing a potential pathway for the synthesis of Ag HNPs with controllable shapes and sizes. These findings not only deepen the understanding of the microscopic formation process of nanosheets, but also elucidate the origin of the observed reversible shape changes, providing new insights into the rational design of controllable nanocrystal shapes in the future, as well as key prerequisites for the understanding of the growth mechanism of nanomaterials and the control of shape-dependent properties.

3.4.3 Growth of 2D nanomaterials through gas–solid interactions. Chemical vapour deposition can achieve high-quality, uniform and controllable growth of 2D nanomaterials on large-area substrates and is a key synthetic method for the preparation of 2D nanomaterials. However, the deposition mechanism of different atoms during the synthesis process and the gas–solid growth mechanism of 2D layered nanomaterials still need to be explored and investigated in depth.[229] Among them, graphene, as a 2D material with high carrier mobility at room temperature, is considered to be one of the most promising candidates for next-generation electronic devices. Researchers investigated the growth mechanism of graphene on different substrates (Cu,[230] Ni,[231] and SiC[232]) by *in situ* TEM. Liu *et al.* used aberration-corrected ambient TEM (AC-ETEM) to observe in real time the nucleation and growth of graphene on an atomic scale Cu substrate (Fig. 12A),[230] and the nucleation of graphene nuclei from the amorphous was observed under a CO/C$_2$H$_4$ atmosphere, carbon atoms for nucleation and growth with gradual ordering of in-plane carbon atoms (Fig. 12B). In addition, the growth of graphene on the edge of copper has a unique lateral epitaxial growth process as well as a step-flow process under a CO/CO$_2$ atmosphere (Fig. 12C). Kling *et al.*[231] found that during the growth of layered carbon structures on nickel (Ni) substrates using acetylene as a carbon precursor, the growth of
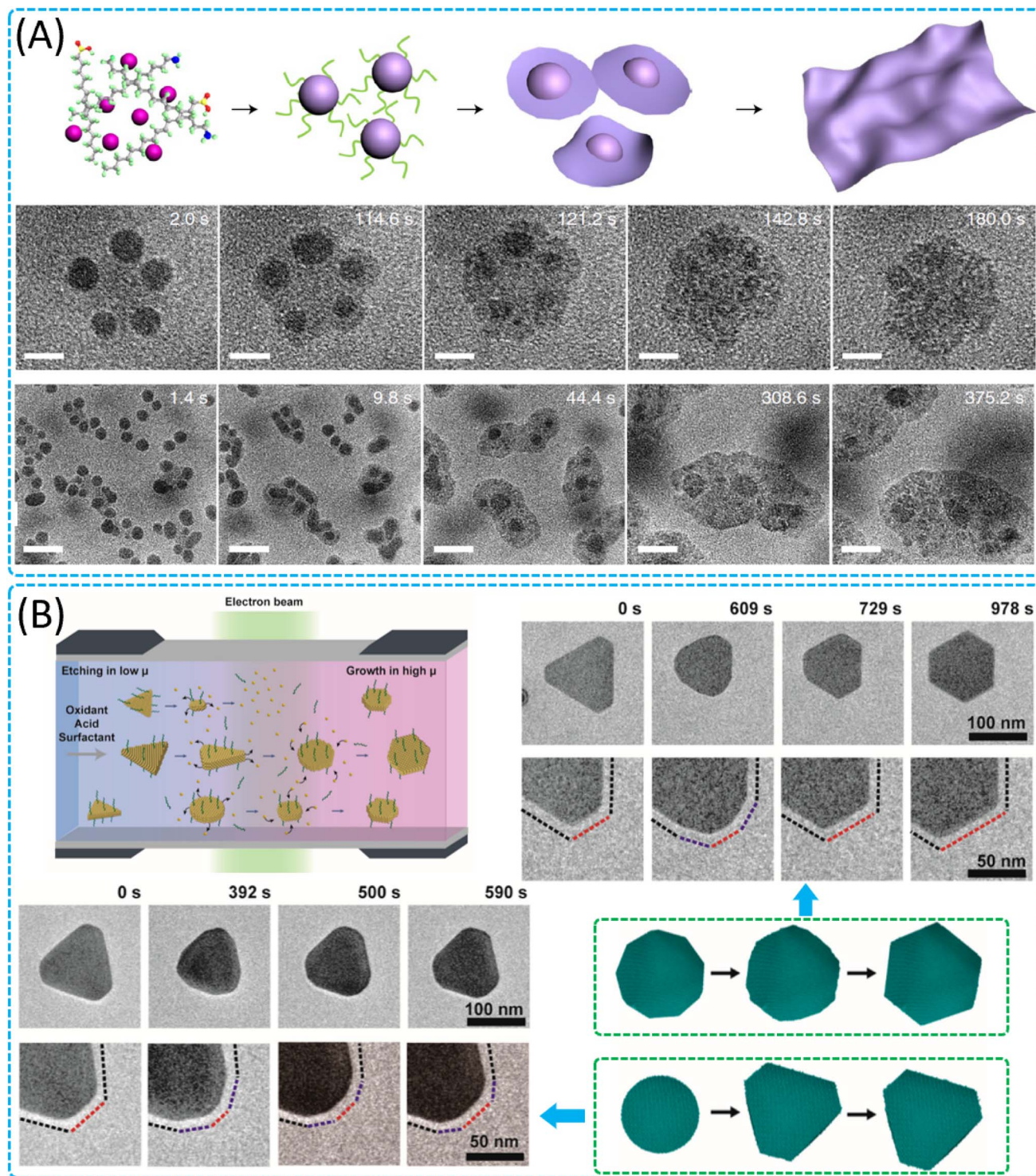
Fig. 11 Growth and shape transformation of 2D nanomaterials in the liquid phase. (A) Schematic showing the formation of a 2D nanosheet from a molecular precursor solution with the pathway of 3D nanoparticle growth and subsequent 3D-to-2D transformations. Sequential images show a few cobalt nickel oxide nanoparticles transforming into a 2D nanosheet and the formation of cobalt nickel oxide nanosheets through the growth of 3D nanoparticles and 3D-to-2D transformations. Reproduced with permission.[215] Copyright 2019, Springer Nature. (B) Schematic illustration of an LPTEM cell for observing diverse shape transformations of Au nanocrystals. Reproduced with permission.[226] Copyright 2023, American Chemical Society.

graphene occurred in two phases by observing the growth of individual graphene layers on the Ni surface, with an initial fast growth phase that was not strongly pressure dependent, followed by a much slower growth phase that was strongly pressure dependent. Meanwhile, Yu *et al.*[232] observed the growth of graphene on SiC surfaces by combining *in situ* TEM at
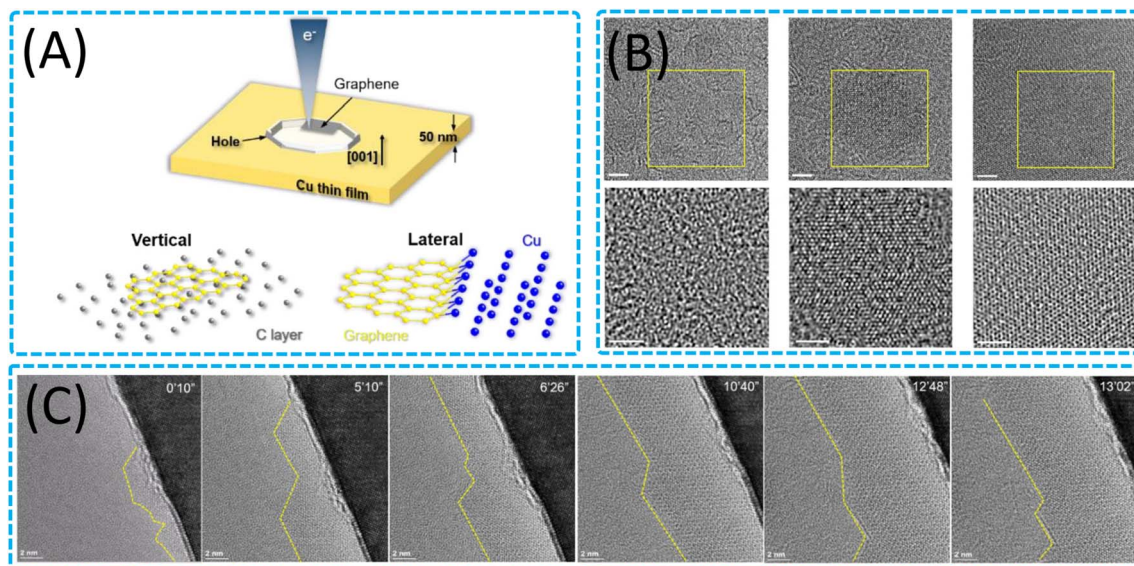
Fig. 12 Growth of 2D nanomaterials through gas−solid interactions. (A) ETEM experimental setup and typical observations of *in situ* graphene growth. (B) Nucleation and growth of graphene from an amorphous C layer. (C) Time-resolved HRTEM images showing the lateral epitaxial growth of graphene on the Cu edge. Reproduced with permission.[230] Copyright 2020, American Chemical Society.

1000 °C at the atomic scale. The results show the sequential decomposition of three SiC layers to form graphene. Sublimation of the first layer leads to the formation of carbon clusters containing short chains and hexagonal rings, and these can be considered as the core of graphene growth. The decomposition of the second layer leads to the joining of new chains with already formed clusters to form a network with large pores. Eventually, the release of carbon atoms from the third layer leads to the disappearance of chains and macropores from the network, forming a complete graphene layer. This study provides a clear picture for understanding the epitaxial growth mechanism of monolayer graphene on SiC. These atomic-scale real-time observations and analyses provide direct evidence for understanding the growth mechanism of graphene and can be extended to other 2D materials.
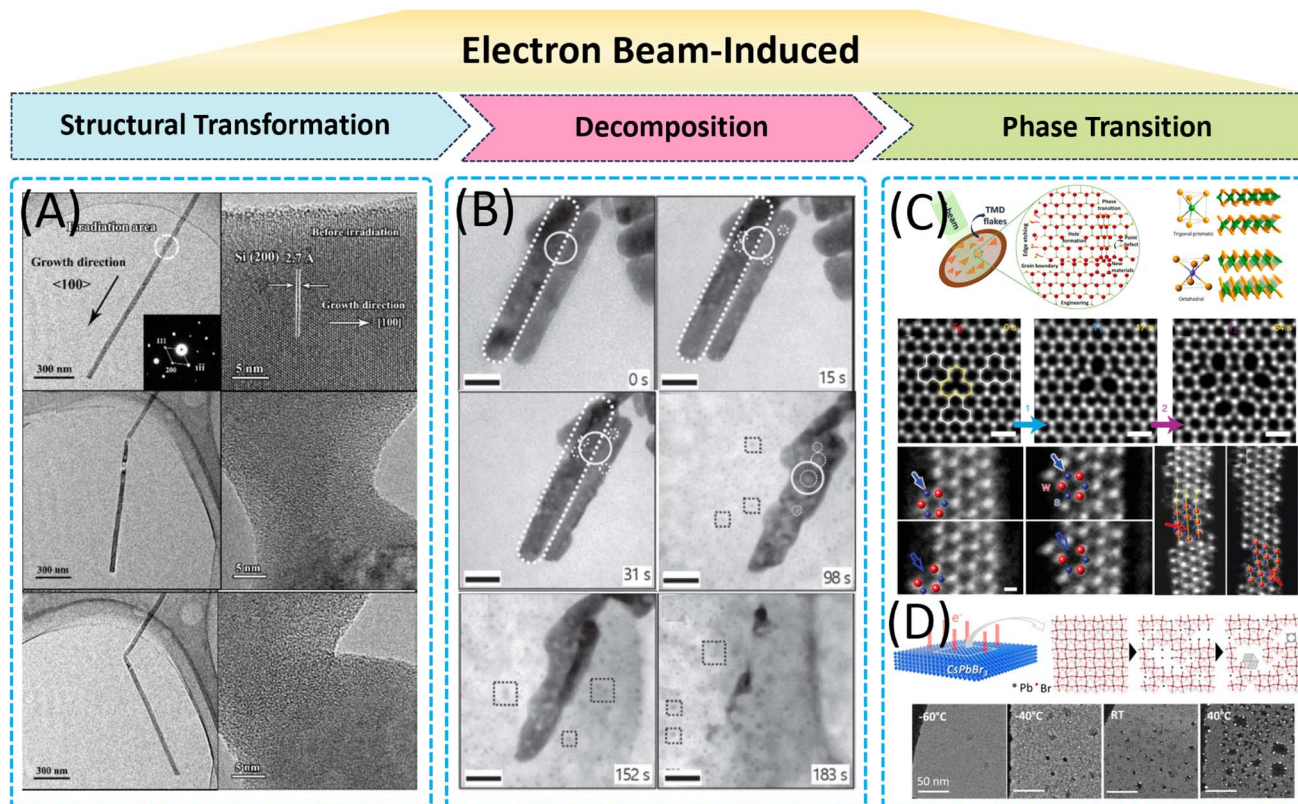
4 Electron-beam induced synthesis for nanomaterials

Electron beams play a pivotal role in the synthesis of nanostructures, offering a versatile and precise method for manipulating materials at the nanoscale.[233–236] The high energy of electron beams allows them to induce a range of physical and chemical changes in materials, such as knock-on displacement,[237] sputtering,[238] and radiolysis,[239] which are essential for nanostructure synthesis. These interactions can lead to the formation of nanoparticles,[233,240] nanowires,[241] nanosheets,[234,242,243] and more complex geometries, all through *in situ* processes within TEM. One significant advantage of using electron beams for nanostructure synthesis is the ability to control the process with atomic precision, which can directly affect the final properties of the nanostructures. TEM provides real-time temporal resolution, enabling researchers to observe

the microstructural evolution as nanostructures are synthesized. On the other hand, electron-beam-induced synthesis can be conducted without specialized specimen holders or peripheral equipment, making it a simpler and rapidly growing approach. It discusses various protocols for synthesizing different dimensional nanostructures, including 0D nanoparticles,[235] 1D nanowires/nanotubes,[244] 2D films,[238] and other exotic geometries like nano-trees or nano-dendrites.[245]

There has been a lot of research focus on the investigation of nanomaterials and their transformations under electron beam irradiation, utilizing advanced TEM techniques. These studies explore the structural changes,[234,243] decomposition,[239,246] and phase transitions[247–250] processes of nanomaterials with different morphologies such as 0D, 1D and 2D materials. Zhu *et al.* investigated the effects of electron-beam irradiation on the structural transformation of silicon (Si) and zinc oxide (ZnO) nanowires. They revealed that electron-beam irradiation can induce a crystal-to-amorphous transition in Si nanowires and surface reconstruction in ZnO nanowires (Fig. 13A).[244] These transformations demonstrate the potential for localized modification of one-dimensional nanomaterials using electron beams. Kim *et al.* observed the rapid decomposition of Bi_2S_3 under electron beam irradiation in water (Fig. 13B),[246] providing insights into the stability and potential applications of such photocatalysts in addressing energy and environmental issues. Meanwhile, more and more researchers are focusing on the transformative effects of electron beam irradiation on nanomaterials, specifically on transition metal dichalcogenides (Fig. 13C)[46] and cesium lead halide perovskites (Fig. 13D).[249] Mendes *et al.*[46] and Manna *et al.*[249] employed TEM to investigate the structural and compositional changes induced by electron irradiation. They explored how electron beams can stimulate desorption of atoms, induce phase transformations, and trigger

Fig. 13 Electron-beam induced synthesis of nanomaterials: electron-beam-induced (A) *in situ* structural transformation in 1D nanomaterials. Reproduced with permission.[244] Copyright 2015, Science China Press. (B) Decomposition of Bi_2S_3 nanorods in water. Reproduced with permission.[246] Copyright 2021, IOP. Phase transition of (C) 2D transition metal dichalcogenides. Reproduced with permission.[46] Copyright 2019, American Chemical Society. (D) Colloidal cesium lead halide perovskite nanocrystals. Reproduced with permission.[249] Copyright 2017, American Chemical Society.

the nucleation and growth of nanoparticles within these materials. The research underscored the potential of electron beam manipulation for material property tuning and the development of nanodevices, while also highlighting the challenges associated with controlling the electron irradiation-induced processes for stable material synthesis and applications in optoelectronics and photovoltaics.

Furthermore, the electron-beam specimen interactions are crucial for understanding the physical background behind the growth mechanisms. The section also discusses the challenges and limitations of using electron beams, such as specimen charging and the risk of inducing uncontrollable structural transformations. The electron-beam-induced synthesis of nanostructures is a powerful technique that leverages the precise interactions between electron beams and materials to create nanostructures with tailored properties.[248] This method is particularly valuable for research and development in nanotechnology, offering a platform for innovation and the potential for large-scale industrial applications once the challenges are addressed.[249]

5   Challenges and conclusion

In situ TEM has proven to be an invaluable tool for investigating the synthesis of nanomaterials at the atomic scale, offering unprecedented insights into nucleation, growth, and structural transformations. However, despite its substantial contributions to the field, there are several challenges associated with *in situ* TEM that need to be addressed to fully harness its potential in nanomaterial synthesis.

5.1   Integration with other techniques

Although *in situ* TEM can provide structural information, real-time chemical characterization at the atomic scale during synthesis is still limited. Firstly, combining *in situ* TEM with advanced spectroscopic methods for real-time chemical analysis is an area that requires further development. Therefore, to gain a comprehensive understanding of nanomaterial synthesis, *in situ* TEM needs to be integrated with other characterization techniques such as spectroscopy, diffraction, and tomography. Secondly, although *in situ* TEM excels at providing vivid, real-time images of nanomaterials' structural and chemical transformations within their chemical milieus, a complete grasp of their synthetic intricacies demands a multifaceted investigational approach. To counter this, an array of innovative *in situ* characterization techniques has swiftly come to the forefront, including but not limited to *in situ* Raman spectroscopy,[251–253] *in situ* infrared spectroscopy,[254–256] *in situ* X-ray diffraction (XRD),[257–259] *in situ* nuclear magnetic resonance
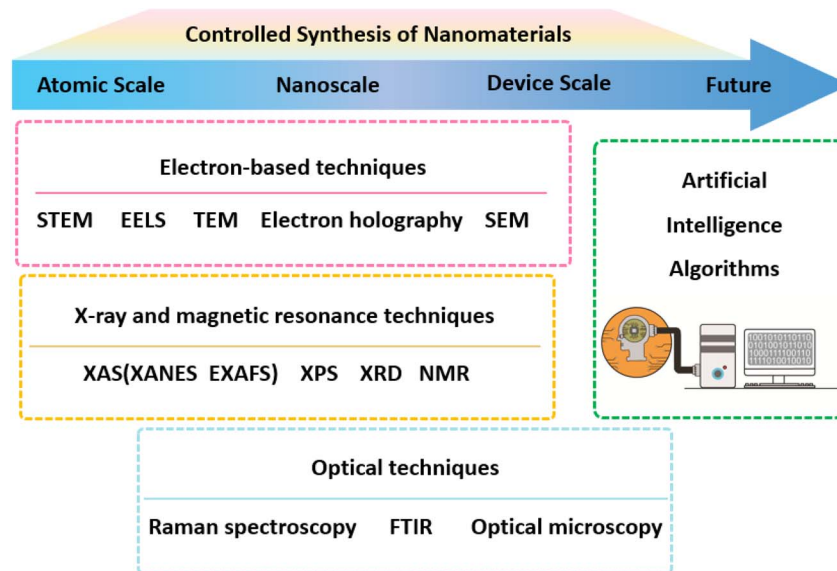
Fig. 14 The most important *in situ* characterization techniques with their spatial resolution scales and the corresponding detection targets in the synthesis of nanomaterials.

(NMR),[260–263] *in situ* X-ray photoelectron spectroscopy (XPS),[264–266] and *in situ* X-ray absorption fine structure (XAFS).[267,268] These advanced techniques have filled the gaps left by conventional approaches. Moreover, the harmonization of these cutting-edge *in situ* methods with established characterization tools is forging new avenues for the invention of novel analytical methodologies. This integrated approach is set to facilitate a more rounded appreciation of nanomaterial synthesis mechanisms, thereby enhancing our ability to design and fabricate materials with tailored properties for specific applications (Fig. 14).[269]

5.2 Complex reaction environments

Replicating realistic synthesis conditions within the high-vacuum environment of a TEM remains a challenge. The need to integrate multiple external stimuli, such as liquid, gas, heat, and light, into a single *in situ* TEM experiment is crucial for mimicking real-world synthesis processes. On the one hand, it is difficult to maintain precise control of the gas and liquid environments within the TEM, including pressure and composition. The spatial and temporal resolution can be compromised due to gas scattering and the need for high-pressure compatibility. On the other hand, accurate temperature measurement at the nanoscale is challenging, particularly when considering the heat effect of the electron beam. This can affect the actual phase transformation temperature and dynamics.

5.3 Data acquisition and analysis

The acquisition of high-quality, high-resolution data in a timely manner is essential for understanding complex nanomaterial synthesis processes. However, the current limitations in data acquisition systems, such as frame rates and image quality, can hinder the detailed analysis of dynamic processes. In addition,

understanding the role of interfaces and compositional changes during nanomaterial synthesis is critical. However, current *in situ* TEM techniques may struggle to provide detailed information on the chemical state, valence, and distribution of elements, particularly light elements, which are often involved in catalytic processes. Hence, the interpretation of *in situ* TEM observations can be complex, particularly when distinguishing between different growth mechanisms or understanding the influence of various reaction parameters. Developing a comprehensive understanding that links observations to underlying mechanisms is an ongoing challenge.

5.4 Temporal and spatial resolution

Capturing the dynamics of nanomaterial synthesis requires high temporal resolution to follow fast processes and high spatial resolution to observe atomic-scale changes. Current *in situ* TEM techniques may not always provide the necessary resolution to capture all relevant details, particularly for very fast or small-scale phenomena.

5.5 Electron beam interaction

The electron beam used in TEM can interact with the sample, causing effects such as heating, knock-on damage, or charging. These interactions can alter the sample's structure and chemistry, potentially leading to observations that do not accurately represent the undisturbed synthesis process.

5.6 Sample preparation and stability

Preparing samples that are representative of actual synthesis conditions and maintaining their stability under electron beams are non-trivial tasks. The need for specialized holders and the potential for sample contamination or damage during preparation and observation add layers of complexity.

Meanwhile, achieving stable loading of nanomaterials within the TEM and maintaining the sample under test conditions without drift is a significant challenge, especially for quantitative nanomechanical tests that require precision.

As mentioned above, the *in situ* TEM method is essential for obtaining high-resolution data on nanocrystal growth in relation to space, time, and energy. We anticipate the development of more intricate *in situ* cultivation settings and a variety of experimental approaches within TEM, including hydrothermal and CVD techniques. Furthermore, the integration of cutting-edge characterization methodologies and advanced data analytics, such as high-throughput experimentation and artificial intelligence algorithms, is anticipated. This synergy will facilitate a more profound comprehension of the underlying nucleation and growth mechanisms of nanocrystals, enabling the meticulous design and crafting of nanocrystals tailored with specific structural and functional attributes. By leveraging these advanced techniques, researchers will gain the ability to elucidate the intricate dynamics of nanocrystal formation with unprecedented clarity, leading to advancements in the precise engineering of materials with customized properties for a wide array of applications. These advances hold immense promise for a wide range of applications in nanomaterials.

Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this review.

Author contributions

All the authors contributed to the literature search, writing, and editing of this review.

Conflicts of interest

The authors declare no conflict of interest.

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References

1 Z. Y. Zhou, N. Tian, J. T. Li, I. Broadwell and S. G. Sun, Nanomaterials of high surface energy with exceptional properties in catalysis and energy storage, *Chem. Soc. Rev.*, 2011, **40**, 4167–4185.

2 X. Qiu, Y. Zhang, Y. Zhu, C. Long, L. Su, S. Liu and Z. Tang, Applications of nanomaterials in asymmetric photocatalysis: recent progress, challenges, and opportunities, *Adv. Mater.*, 2021, **33**, 2001731.

3 E. Pomerantseva, F. Bonaccorso, X. Feng, Y. Cui and Y. Gogotsi, Energy storage: The future enabled by nanomaterials, *Science*, 2019, **366**, eaan8285.

4 X. Yang, M. Yang, B. Pang, M. Vara and Y. Xia, Gold nanomaterials at work in biomedicine, *Chem. Rev.*, 2015, **115**, 10410–10488.

5 X. Wang, X. Zhong, J. Li, Z. Liu and L. Cheng, Inorganic nanomaterials with rapid clearance for biomedical applications, *Chem. Soc. Rev.*, 2021, **50**, 8669–8742.

6 Z. Fan and H. Zhang, Crystal phase-controlled synthesis, properties and applications of noble metal nanomaterials, *Chem. Soc. Rev.*, 2016, **45**, 63–82.

7 N. Baig, I. Kammakakam and W. Falath, Nanomaterials: a review of synthesis methods, properties, recent progress, and challenges, *Mater. Adv.*, 2021, **2**, 1821–1871.

8 N. Abid, A. M. Khan, S. Shujait, K. Chaudhary, M. Ikram, M. Imran, J. Haider, M. Khan, Q. Khan and M. Maqbool, Synthesis of nanomaterials using various top-down and bottom-up approaches, influencing factors, advantages, and disadvantages: A review, *Adv. Colloid Interface Sci.*, 2022, **300**, 102597.

9 N. T. K. Thanh, N. Maclean and S. Mahiddine, Mechanisms of nucleation and growth of nanoparticles in solution, *Chem. Rev.*, 2014, **114**, 7610–7630.

10 J. Lee, J. Yang, S. G. Kwon and T. Hyeon, Nonclassical nucleation and growth of inorganic nanoparticles, *Nat. Rev. Mater.*, 2016, **1**, 1–16.

11 C. Jia, A. Xiao, J. Zhao, P. Wang, X. Fang, H. Zhang and B. Guan, A new perspective on crystal nucleation: a classical view on non-classical nucleation, *Cryst. Growth Des.*, 2024, **24**, 601–612.

12 Y. S. Jun, Y. Zhu, Y. Wang, D. Ghim, X. Wu, D. Kim and H. Jung, Classical and nonclassical nucleation and growth mechanisms for nanoparticle formation, *Annu. Rev. Phys. Chem.*, 2022, **73**, 453–477.

13 M. J. Ansari, M. M. Kadhim, B. A. Hussein, H. A. Lafta and E. Kianfar, Synthesis and stability of magnetic nanoparticles, *J. Bionanosci.*, 2022, **12**, 627–638.

14 B. Rezaei, P. Yari, S. M. Sanders, H. Wang, V. K. Chugh, S. Liang, S. Mostufa, K. Xu, J. P. Wang, J. Gómez-Pastora and K. Wu, Magnetic nanoparticles: a review on synthesis, characterization, functionalization, and biomedical applications, *Small*, 2024, **20**, 2304848.

15 Z. Ma, J. Mohapatra, K. Wei, J. P. Liu and S. Sun, Magnetic nanoparticles: synthesis, anisotropy, and applications, *Chem. Rev.*, 2023, **123**, 3904–3943.

16 Y. Han, L. Wang, K. Cao, J. Zhou, Y. Zhu, Y. Hou and Y. Lu, *In situ* TEM characterization and modulation for phase engineering of nanomaterials, *Chem. Rev.*, 2023, **123**, 14119–14184.

17 J. Wu, H. Shan, W. Chen, X. Gu, P. Tao, C. Song, W. Shang and T. Deng, *In situ* environmental TEM in imaging gas and liquid phase chemical reactions for materials research, *Adv. Mater.*, 2016, **28**, 9686–9712.

18 R. R. Unocic and E. A. Stach, Gas-phase electron microscopy for materials research, *MRS Bull.*, 2023, **48**, 828–832.

19 C. Luo, C. Wang, X. Wu, J. Zhang and J. Chu, *In situ* transmission electron microscopy characterization and manipulation of two-dimensional layered materials beyond graphene, *Small*, 2017, **13**, 1604259.

20 F. M. Alcorn, P. K. Jain and R. M. Van Der Veen, Time-resolved transmission electron microscopy for nanoscale chemical dynamics, *Nat. Rev. Chem.*, 2023, **7**, 256–272.

21 H. Ye, Z. Zhang and R. Wang, Nucleation and growth of nanocrystals investigated by *in situ* transmission electron microscopy, *Small*, 2023, **19**, 2303872.

22 Z. Xu and Z. Ou, Direct imaging of the kinetic crystallization pathway: simulation and liquid-phase transmission electron microscopy observations, *Materials*, 2023, **16**, 2026.

23 Q. Chen, J. M. Yuk, M. R. Hauwiller, J. Park, K. S. Dae, J. S. Kim and A. P. Alivisatos, Nucleation, growth, and superlattice formation of nanocrystals observed in liquid cell transmission electron microscopy, *MRS Bull.*, 2020, **45**, 713–726.

24 H. Y. Chao, K. Venkatraman, S. Moniri, Y. Jiang, X. Tang, S. Dai, W. Gao, J. Miao and M. Chi, *In situ* and emerging transmission electron microscopy for catalysis research, *Chem. Rev.*, 2023, **123**, 8347–8394.

25 B. He, Y. Zhang, X. Liu and L. Chen, *In situ* transmission electron microscope techniques for heterogeneous catalysis, *ChemCatChem*, 2020, **12**, 1853–1872.

26 J. Xie, J. Li, W. Mai and G. Hong, A decade of advanced rechargeable batteries development guided by *in situ* transmission electron microscopy, *Nano Energy*, 2021, **83**, 105780.

27 Y. Li, H. Xu, Q. Ning, S. Li, J. Wang, J. Wang, Z. Hu, J. Tian, X. Li, Y. Han and Y. Zhu, Visualizing structure, growth, and dynamics of Li dendrite in batteries: from atomic to device scales, *Adv. Funct. Mater.*, 2024, **34**, 2401361.

28 H. Zhao, Y. Zhu, H. Ye, Y. He, H. Li, Y. Sun, F. Yang and R. Wang, Atomic-scale structure dynamics of nanocrystals revealed by *in situ* and environmental transmission electron microscopy, *Adv. Mater.*, 2023, **35**, 2206911.

29 J. Zhang, M. Li, Z. Kang, B. Xiao, H. Lin, J. Lu, H. Liu, X. Zhang, D. L. Peng and Q. Zhang, Atomic mechanisms of hexagonal close-packed Ni nanocrystallization revealed by *in situ* liquid cell transmission electron microscopy, *Nano Res.*, 2022, **15**, 6772–6778.

30 H. Yoshida, S. Takeda, T. Uchiyama, H. Kohno and Y. Homma, Atomic-scale *in situ* observation of carbon nanotube growth from solid state iron carbide nanoparticles, *Nano Lett.*, 2008, **8**, 2082–2086.

31 Y. Wang, L. Qiu, L. Zhang, D. M. Tang, R. Ma, Y. Wang, B. Zhang, F. Ding, C. Liu and H. M. Cheng, Precise identification of the active phase of cobalt catalyst for carbon nanotube growth by *in situ* transmission electron microscopy, *ACS Nano*, 2020, **14**, 16823–16831.

32 L. Tang, W. Wu, L. He, K. Yu, T. Xu, Q. Zhang, L. Zhang and L. Sun, Novel interface in CuAg nanostructure induced by size effect, *J. Phys. Chem. Lett.*, 2019, **10**, 1973–1980.

33 Z. Zeng, X. Zhang, K. Bustillo, K. Niu, C. Gammer, J. Xu and H. Zheng, *In situ* study of lithiation and delithiation of MoS_2 nanosheets using electrochemical liquid cell transmission electron microscopy, *Nano Lett.*, 2015, **15**, 5214–5220.

34 R. Yang, L. Mei, Y. Fan, Q. Zhang, H. G. Liao, J. Yang, J. Li and Z. Zeng, Fabrication of liquid cell for *in situ* transmission electron microscopy of electrochemical processes, *Nat. Protoc.*, 2023, **18**, 555–578.

35 N. Hodnik, G. Dehm and K. J. J. Mayrhofer, Importance and challenges of electrochemical *in situ* liquid cell electron microscopy for energy conversion research, *Acc. Chem. Res.*, 2016, **49**, 2015–2022.

36 J. Park, K. Koo, N. Noh, J. H. Chang, J. Y. Cheong, K. S. Dae, J. S. Park, S. Ji, I. D. Kim and J. M. Yuk, Graphene liquid cell electron microscopy: progress, applications, and perspectives, *ACS Nano*, 2021, **15**, 288–308.

37 D. J. Kelly, M. Zhou, N. Clark, M. J. Hamer, E. A. Lewis, A. M. Rakowski, S. J. Haigh and R. V. Gorbachev, Nanometer resolution elemental mapping in graphene-based TEM liquid cells, *Nano Lett.*, 2018, **18**, 1168–1174.

38 J. M. Yuk, H. K. Seo, J. W. Choi and J. Y. Lee, Anisotropic lithiation onset in silicon nanoparticle anode revealed by *in situ* graphene liquid cell electron microscopy, *ACS Nano*, 2014, **8**, 7478–7485.

39 F. Ye, M. Xu, S. Dai, P. Tieu, X. Ren and X. Pan, *In situ* TEM studies of catalysts using windowed gas cells, *Catalysts*, 2020, **10**, 779.

40 F. Wu and N. Yao, Advances in windowed gas cells for *in situ* TEM studies, *Nano Energy*, 2015, **13**, 735–756.

41 L. Zhang, T. Yang, C. Du, Q. Liu, Y. Tang, J. Zhao, B. Wang, T. Chen, Y. Sun, P. Jia, H. Li, L. Geng, J. Chen, H. Ye, Z. Wang, Y. Li, H. Sun, X. Li, Q. Dai, Y. Tang, Q. Peng, T. Shen, S. Zhang, T. Zhu and J. Huang, Lithium whisker growth and stress generation in an *in situ* atomic force microscope-environmental transmission electron microscope set-up, *Nat. Nanotechnol.*, 2020, **15**, 94–98.

42 J. R. Jinschek, Advances in the environmental transmission electron microscope (ETEM) for nanoscale *in situ* studies of gas-solid interactions, *Chem. Commun.*, 2014, **50**, 2696–2706.

43 D. Su, F. Wang, C. Ma and N. Jiang, Engineering nano-composite $Li_4Ti_5O_{12}$ anodes via scanning electron-probe fabrication, *Nano Energy*, 2013, **2**, 343–350.

44 M. Xu, T. Deng, C. Li, H. Zhao, J. Wang, Y. Liu, J. Wang, G. Feng, N. Li, S. Ding and K. Xi, Oxygen deficient Eu_2O_3-synchronizes the shielding and catalytic conversion of polysulfides toward high-performance lithium sulfur batteries, *Chin. Chem. Lett.*, 2024, 110372.

45 N. Li, Y. Zhang, H. Zhao, Z. Liu, X. Zhang and Y. Du, Synthesis of high-quality α-MnSe nanostructures with superior lithium storage properties, *Inorg. Chem.*, 2016, **55**, 2765–2770.

46 R. G. Mendes, J. Pang, A. Bachmatiuk, H. Q. Ta, L. Zhao, T. Gemming, L. Fu, Z. Liu and M. H. Rümmeli, Electron-driven *in situ* transmission electron microscopy of 2D transition metal dichalcogenides and their 2D heterostructures, *ACS Nano*, 2019, **13**, 978–995.

47 R. Podor, V. Trillaud, G. I. N. Bouala, N. Dacheux, C. Ricolleau and N. Clavier, A multiscale *in situ* high temperature high resolution transmission electron microscopy study of ThO$_2$ sintering, *Nanoscale*, 2021, **13**, 7362–7374.

48 A. Chauvin, L. Molina-Luna, J. Ding, C. H. Choi, P. Y. Tessier and A. A. El Mel, Study of the coarsening of nanoporous gold nanowires by *in situ* scanning transmission electron microscopy during annealing, *Phys. Status Solidi RRL*, 2019, **13**, 1900376.

49 Y. Lee, J. Lee, H. Chung, J. Kim and Z. Lee, *In situ* scanning transmission electron microscopy study of MoS$_2$ formation on graphene with a deep-learning framework, *ACS Omega*, 2021, **6**, 21623–21630.

50 D. S. Gavhane, A. D. Sontakke and M. A. van Huis, Thermolysis-driven growth of vanadium oxide nanostructures revealed by *in situ* transmission electron microscopy: implications for battery applications, *ACS Appl. Nano Mater.*, 2023, **6**, 7280–7289.

51 L. Fei, S. Lei, W. B. Zhang, W. Lu, Z. Lin, C. H. Lam, Y. Chai and Y. Wang, Direct TEM observations of growth mechanisms of two-dimensional MoS$_2$ flakes, *Nat. Commun.*, 2016, **7**, 12206.

52 M. Xiao, H. Sun, Y. Meng and F. Zhu, Advances of *in situ* transmission electron microscopy research on gas phase catalyst particles, *Catal. Sci. Technol.*, 2024, **14**, 2040–2063.

53 Y. Chen, K. Yin, T. Xu, H. Guo and L. Sun, Characterization of nanomaterials using *in situ* liquid-cell transmission electron microscopy: a review, *ACS Appl. Nano Mater.*, 2023, **6**, 22545–22567.

54 S. Pu, C. Gong and A. W. Robertson, Liquid cell transmission electron microscopy and its applications, *R. Soc. Open Sci.*, 2020, **7**, 191204.

55 Q. Cheek, E. Fahrenkrug, S. Hlynchuk, D. H. Alsem, N. J. Salmon and S. Maldonado, *In situ* transmission electron microscopy measurements of Ge nanowire synthesis with liquid metal nanodroplets in water, *ACS Nano*, 2020, **14**, 2869–2879.

56 I. A. Moreno-Hernandez, M. F. Crook, V. Jamali and A. P. Alivisatos, Recent advances in the study of colloidal nanocrystals enabled by *in situ* liquid-phase transmission electron microscopy, *MRS Bull.*, 2022, **47**, 305–313.

57 Y. Sun, Controlled synthesis of colloidal silver nanoparticles in organic solutions: empirical rules for nucleation engineering, *Chem. Soc. Rev.*, 2013, **42**, 2497–2511.

58 V. K. LaMer and R. H. Dinegar, Theory, production and mechanism of formation of monodispersed hydrosols, *J. Am. Chem. Soc.*, 1950, **72**, 4847–4854.

59 D. Erdemir, A. Y. Lee and A. S. Myerson, Nucleation of crystals from solution: classical and two-step models, *Acc. Chem. Res.*, 2009, **42**, 621–629.

60 S. Jeon, T. Heo, S. Y. Hwang, J. Ciston, K. C. Bustillo, B. W. Reed, J. Ham, S. Kang, S. Kim, J. Lim, K. Lim, J. S. Kim, M. H. Kang, R. S. Bloom, S. Hong, K. Kim, A. Zettl, W. Y. Kim, P. Ercius, J. Park and W. C. Lee, Reversible disorder-order transitions in atomic crystal nucleation, *Science*, 2021, **371**, 498–503.

61 K. A. Altammar, A review on nanoparticles: characteristics, synthesis, applications, and challenges, *Front. Microbiol.*, 2023, **14**, 1155622.

62 J. Park, T. Kwon, J. Kim, H. Jin, H. Y. Kim, B. Kim, S. H. Joo and K. Lee, Hollow nanoparticles as emerging electrocatalysts for renewable energy conversion reactions, *Chem. Soc. Rev.*, 2018, **47**, 8173–8202.

63 A. Fereydooni, C. Yue and Y. Chao, A brief overview of silicon nanoparticles as anode material: a transition from lithium-ion to sodium-ion batteries, *Small*, 2024, **20**, 2307275.

64 R. Saha, B. Mondal and P. S. Mukherjee, Molecular cavity for catalysis and formation of metal nanoparticles for use in catalysis, *Chem. Rev.*, 2022, **122**, 12244–12307.

65 D. Astruc, Introduction: nanoparticles in catalysis, *Chem. Rev.*, 2020, **120**, 461–463.

66 M. S. Rider, Á. Buendía, D. R. Abujetas, P. A. Huidobro, J. A. Sánchez-Gil and V. Giannini, Advances and prospects in topological nanoparticle photonics, *ACS Photonics*, 2022, **9**, 1483–1499.

67 E. C. Wang and A. Z. Wang, Nanoparticles and their applications in cell and molecular biology, *Integr. Biol.*, 2014, **6**, 9–26.

68 W. J. Stark, Nanoparticles in biological systems, *Angew. Chem., Int. Ed.*, 2011, **50**, 1242–1258.

69 M. Notarianni, K. Vernon, A. Chou, M. Aljada, J. Liu and N. Motta, Plasmonic effect of gold nanoparticles in organic solar cells, *Sol. Energy*, 2014, **106**, 23–37.

70 D. Garcia Romero, L. Di Mario, F. Yan, C. M. Ibarra-Barreno, S. Mutalik, L. Protesescu, P. Rudolf and M. A. Loi, Understanding the surface chemistry of SnO$_2$ nanoparticles for high performance and stable organic solar cells, *Adv. Funct. Mater.*, 2024, **34**, 2307958.

71 G. Zhang, Q. Chen, Z. Zhang, Z. Gao, C. Xiao, Y. Wei and W. Li, NiO$_x$ nanoparticles hole-transporting layer regulated by ionic radius-controlled doping and reductive agent for organic solar cells with efficiency of 19.18%, *Adv. Mater.*, 2024, **36**, 2310630.

72 H. Wu, W. Yan, Y. Xing, L. Li, J. Liu, L. Li, P. Huang, C. Lai, C. Wang, W. Chen and S. Chou, Tailoring the interfacial electric field using silicon nanoparticles for stable zinc-ion batteries, *Adv. Funct. Mater.*, 2024, **34**, 2213882.

73 Y. Liu, F. Bettels, Z. Lin, Z. Li, Y. Shao, F. Ding, S. Liu and L. Zhang, Recent advances in transition-metal-based catalytic material for room-temperature sodium-sulfur batteries, *Adv. Funct. Mater.*, 2024, **34**, 2302626.

74 M. Khan, S. Yan, M. Ali, F. Mahmood, Y. Zheng, G. Li, J. Liu, X. Song and Y. Wang, Innovative solutions for high-performance silicon anodes in lithium-ion batteries: overcoming challenges and real-world applications, *Nano-Micro Lett.*, 2024, **16**, 179.

75 M. Cai, C. Li, X. An, B. Zhong, Y. Zhou, K. Feng, S. Wang, C. Zhang, M. Xiao, Z. Wu, J. He, C. Wu, J. Shen, Z. Zhu, K. Feng, J. Zhong and L. He, Supra-photothermal CO$_2$ methanation over greenhouse-like plasmonic

superstructures of ultrasmall cobalt nanoparticles, *Adv. Mater.*, 2024, **36**, 2308859.

76 I. Venditti, Gold nanoparticles in photonic crystals applications: a review, *Materials*, 2017, **10**, 97.

77 T. An, X. Jiang, F. Gao, C. Schäfer, J. Qiu, N. Shi, X. Song, M. Zhang, C. E. Finlayson, X. Zheng, X. Li, F. Tian, B. Zhu, T. Sui, X. Han, J. J. Baumberg, T. Fan and Q. Zhao, Strain to shine: stretching-induced three-dimensional symmetries in nanoparticle-assembled photonic crystals, *Nat. Commun.*, 2024, **15**, 5215.

78 Y. Hang, A. Wang and N. Wu, Plasmonic silver and gold nanoparticles: shape and structure-modulated plasmonic functionality for point-of-caring sensing, bio-imaging and medical therapy, *Chem. Soc. Rev.*, 2024, **53**, 2932–2971.

79 R. Ghosh Chaudhuri and S. Paria, Core/shell nanoparticles: classes, properties, synthesis mechanisms, characterization, and applications, *Chem. Rev.*, 2012, **112**, 2373–2433.

80 S. Anu Mary Ealia and M. P. Saravanakumar, A review on the classification, characterisation, synthesis of nanoparticles and their application, *IOP Conf. Ser.: Mater. Sci. Eng.*, 2017, **263**, 032019.

81 J. Kim, S. Kang, F. Cheng, Y. Wang, X. Ye and J. Park, Recent advances in liquid phase transmission electron microscopy of nanoparticle growth and self-assembly, *MRS Bull.*, 2024, **49**, 365–376.

82 B. H. Kim, J. Yang, D. Lee, B. K. Choi, T. Hyeon and J. Park, Liquid-phase transmission electron microscopy for studying colloidal inorganic nanoparticles, *Adv. Mater.*, 2018, **30**, 1703316.

83 A. Chen, T. U. Dissanayake, J. Sun and T. J. Woehl, Unraveling chemical processes during nanoparticle synthesis with liquid phase electron microscopy and correlative techniques, *Chem. Commun.*, 2023, **59**, 12830–12846.

84 H. J. Han, G. R. Lee, Y. Xie, H. Jang, D. J. Hynek, E. N. Cho, Y. J. Kim, Y. S. Jung and J. J. Cha, Unconventional grain growth suppression in oxygen-rich metal oxide nanoribbons, *Sci. Adv.*, 2021, **7**, eabh2012.

85 W. Gao, P. Tieu, C. Addiego, Y. Ma, J. Wu and X. Pan, Probing the dynamics of nanoparticle formation from a precursor at atomic resolution, *Sci. Adv.*, 2019, **5**, eaau9590.

86 T. M. Diallo, M. R. Aziziyan, R. Arvinte, J. C. Harmand, G. Patriarche, C. Renard, S. Fafard, R. Arès and A. Boucherif, *In situ* transmission electron microscopy observation of germanium growth on freestanding graphene: unfolding mechanism of 3D crystal growth during van der waals epitaxy, *Small*, 2022, **18**, 2101890.

87 L. Fei, S. M. Ng, W. Lu, M. Xu, L. Shu, W. B. Zhang, Z. Yong, T. Sun, C. H. Lam, C. W. Leung, C. L. Mak and Y. Wang, Atomic-scale mechanism on nucleation and growth of Mo_2C nanoparticles revealed by *in situ* transmission electron microscopy, *Nano Lett.*, 2016, **16**, 7875–7881.

88 V. Sudheeshkumar, C. Soong, S. Dogel and R. W. J. Scott, Probing the thermal stability of (3-mercaptopropyl)-trimethoxysilane-protected Au25 clusters by *in situ*

89 T. R. Henninen, M. Bon, F. Wang, D. Passerone and R. Erni, The structure of sub-nm platinum clusters at elevated temperatures, *Angew. Chem., Int. Ed.*, 2020, **59**, 839–845.

90 H. M. Ashberry, X. Zhan and S. E. Skrabalak, Identification of nanoscale processes associated with the disorder-to-order transformation of carbon-supported alloy nanoparticles, *ACS Mater. Au*, 2022, **2**, 143–153.

91 Y. Zhou, X. Deng, H. Xing, H. Zhao, Y. Liu, L. Guo, J. Feng, W. Feng, Y. Zong, X. Zhu, X. Li, Y. Peng and X. Zheng, Dynamically observing the formation of MOFs-driven Co/N-doped carbon nanocomposites by *in situ* transmission electron microscope and their application as high-efficient microwave absorbent, *Nano Res.*, 2022, **15**, 6819–6830.

92 M. M. Rusu, A. Vulpoi, I. Maurin, L. C. Cotet, L. C. Pop, C. I. Fort, M. Baia, L. Baia and I. Florea, Thermal evolution of C-Fe-Bi nanocomposite system: from nanoparticle formation to heterogeneous graphitization stage, *Microsc. Microanal.*, 2022, **28**, 317–329.

93 R. Bergamaschini, B. A. Rosen, F. Montalenti and J. Colin, Motion of crystalline inclusions by interface diffusion in the proximity of free surfaces, *J. Nanopart. Res.*, 2019, **21**, 271.

94 F. Li, Y. Zong, Y. Ma, M. Wang, W. Shang, P. Tao, C. Song, T. Deng, H. Zhu and J. Wu, Atomistic imaging of competition between surface diffusion and phase transition during the intermetallic formation of faceted particles, *ACS Nano*, 2021, **15**, 5284–5293.

95 Y. Shimada, Y. Ikeda, K. Yoshida, M. Sato, J. Chen, Y. Du, K. Inoue, R. Maaß, Y. Nagai and T. J. Konno, *In situ* thermal annealing transmission electron microscopy of irradiation induced Fe nanoparticle precipitation in Fe-Si alloy, *J. Appl. Phys.*, 2022, **131**, 164902.

96 M. Z. Hussain, M. Bahri, W. R. Heinz, Q. Jia, O. Ersen, T. Kratky, R. A. Fischer, Y. Zhu and Y. Xia, An *in situ* investigation of the thermal decomposition of metal-organic framework NH_2-MIL-125 (Ti), *Microporous Mesoporous Mater.*, 2021, **316**, 110957.

97 L. Su, X. Chen, L. Xu, T. Eldred, J. Smith, C. DellaRova, H. Wang and W. Gao, Visualizing the formation of high-entropy fluorite oxides from an amorphous precursor at atomic resolution, *ACS Nano*, 2022, **16**, 21397–21406.

98 M. Gocyla, S. Kuehl, M. Shviro, H. Heyen, S. Selve, R. E. Dunin-Borkowski, M. Heggen and P. Strasser, Shape stability of octahedral PtNi nanocatalysts for electrochemical oxygen reduction reaction studied by *in situ* transmission electron microscopy, *ACS Nano*, 2018, **12**, 5306–5311.

99 Z. Lyu, R. Chen, M. Mavrikakis and Y. Xia, Physical transformations of noble-metal nanocrystals upon thermal activation, *Acc. Chem. Res.*, 2021, **54**, 1–10.

100 M. Zhao, Z. Chen, Z. Lyu, Z. D. Hood, M. Xie, M. Vara, M. Chi and Y. Xia, Ru octahedral nanocrystals with a face-centered cubic structure, {111} facets, thermal

stability up to 400 °C, and enhanced catalytic activity, *J. Am. Chem. Soc.*, 2019, **141**, 7028–7036.

101 M. Gatalo, F. Ruiz-Zepeda, N. Hodnik, G. Dražić, M. Bele and M. Gaberšček, Insights into thermal annealing of highly-active PtCu₃/C oxygen reduction reaction electrocatalyst: an *in situ* heating transmission electron microscopy study, *Nano Energy*, 2019, **63**, 103892.

102 X. Li, Y. He, S. Cheng, B. Li, Y. Zeng, Z. Xie, Q. Meng, L. Ma, K. Kisslinger, X. Tong, S. Hwang, S. Yao, C. Li, Z. Qiao, C. Shan, Y. Zhu, J. Xie, G. Wang, G. Wu and D. Su, Atomic structure evolution of Pt-Co binary catalysts: single metal sites versus intermetallic nanocrystals, *Adv. Mater.*, 2021, **33**, 2106371.

103 X. Li, S. Cheng, Y. He, L. Qian, D. Zakharov, G. Wu, C. Shan, L. Zhang and D. Su, Revealing the dynamics of the alloying and segregation of Pt-Co nanoparticles via *in situ* environmental transmission electron microscopy, *Nano Res.*, 2023, **16**, 3055–3062.

104 L. Tang, W. Wu, L. He, T. Xu, H. Dong, L. Zhang, L. Shi and L. Sun, *In situ* observation of the solid solution-induced sublimation of CuAg Janus nanoparticles, *J. Alloys Compd.*, 2021, **877**, 160168.

105 L. Tang, W. Wu, L. He, K. Yu, T. Xu, Q. Zhang, L. Zhang and L. Sun, Novel interface in CuAg nanostructure induced by size effect, *J. Phys. Chem. Lett.*, 2019, **10**, 1973–1980.

106 M. A. Asoro, D. Kovar and P. J. Ferreira, *In situ* transmission electron microscopy observations of sublimation in silver nanoparticles, *ACS Nano*, 2013, **7**, 7844–7852.

107 N. D. Loh, S. Sen, M. Bosman, S. F. Tan, J. Zhong, C. A. Nijhuis, P. Král, P. Matsudaira and U. Mirsaidov, Multistep nucleation of nanocrystals in aqueous solution, *Nat. Chem.*, 2017, **9**, 77–82.

108 W. Dachraoui and R. Erni, Nonclassical nucleation and growth of Pd nanocrystals from aqueous solution studied by *in situ* liquid transmission electron microscopy, *Chem. Mater.*, 2023, **35**, 1201–1208.

109 T. J. Woehl, Metal nanocrystal formation during liquid phase transmission electron microscopy: thermodynamics and kinetics of precursor conversion, nucleation, and growth, *Chem. Mater.*, 2020, **32**, 7569–7581.

110 B. Mei, A. J. Moreno and K. S. Schweizer, Unified understanding of the structure, thermodynamics, and diffusion of single-chain nanoparticle fluids, *ACS Nano*, 2024, **18**, 15529–15544.

111 M. Borowska, K. Jankowski, M. Trzaskowski, W. Wojtasiak, P. Korpas, S. Kozłowski and D. Gryglewski, Nucleation and growth of zinc-templated mesoporous selenium nanoparticles and potential non-thermal effects during their microwave-assisted synthesis, *Sci. Rep.*, 2024, **14**, 31444.

112 Y. Sun, X. Zhang, R. Huang, D. Yang, J. Kim, J. Chen, E. H. Ang, M. Li, L. Li and X. Song, Revealing microscopic dynamics: *in situ* liquid-phase TEM for live observations of soft materials and quantitative analysis via deep learning, *Nanoscale*, 2024, **16**, 2945–2954.

113 J. H. Nam, G. Nayak, S. Exarhos, C. M. Mueller, D. Xu, G. C. Schatz and P. J. Bruggeman, Mechanisms of controlled stabilizer-free synthesis of gold nanoparticles in liquid aerosol containing plasma, *Chem. Sci.*, 2024, **15**, 11643–11656.

114 R. G. Finke, M. A. Watzky and C. B. Whitehead, Response to "Particle size is a primary determinant for sigmoidal kinetics of nanoparticle formation: a 'disproof' of the finke-watzky (F-W) nanoparticle nucleation and growth mechanism.", *Chem. Mater.*, 2020, **32**, 3657–3672.

115 Y. S. Pestovsky and T. Srichana, Formation of aggregate-free gold nanoparticles in the cyclodextrin-tetrachloroaurate system follows finke-watzky kinetics, *Nanomaterials*, 2022, **12**, 583.

116 D. Keller, T. R. Henninen and R. Erni, Atomic mechanisms of gold nanoparticle growth in ionic liquids studied by *in situ* scanning transmission electron microscopy, *Nanoscale*, 2020, **12**, 22511–22517.

117 S. Y. Hou, C. Y. Huang, S. B. Tsai, J. Y. Chen and W. W. Wu, *In situ* atomic-scale TEM observation of Ag nanoparticle-mediated coalescence in liquids, *Appl. Surf. Sci.*, 2021, **546**, 149057.

118 Z. Aabdin, J. Lu, X. Zhu, U. Anand, N. D. Loh, H. Su and U. Mirsaidov, Bonding pathways of gold nanocrystals in solution, *Nano Lett.*, 2014, **14**, 6639–6643.

119 T. Su, Z. L. Wang and Z. Wang, *In situ* observations of shell growth and oxidative etching behaviors of Pd nanoparticles in solutions by liquid cell transmission electron microscopy, *Small*, 2019, **15**, 1900050.

120 W. Gao, A. O. Elnabawy, Z. D. Hood, Y. Shi, X. Wang, L. T. Roling, X. Pan, M. Mavrikakis, Y. Xia and M. Chi, Atomistic insights into the nucleation and growth of platinum on palladium nanocrystals, *Nat. Commun.*, 2021, **12**, 3215.

121 J. Wu, W. Gao, J. Wen, D. J. Miller, P. Lu, J. M. Zuo and H. Yang, Growth of Au on Pt icosahedral nanoparticles revealed by low-dose *in situ* TEM, *Nano Lett.*, 2015, **15**, 2711–2715.

122 W. Wei, T. Bai, R. Fu, L. Sun, W. Wang, M. Dong, L. Chen, Z. Guo and F. Xu, Unravelling the shell growth pathways of Au-Ag core-shell nanoparticles by *in situ* liquid cell transmission electron microscopy, *Nanoscale*, 2021, **13**, 3136–3143.

123 S. B. Tsai, J. Y. Chen, C. Y. Huang, S. Y. Hou and W. W. Wu, Observing growth and crystallization of Au@ZnO core-shell nanoparticles by *in situ* liquid cell transmission electron microscopy: implications for photocatalysis and gas-sensing applications, *ACS Appl. Nano Mater.*, 2021, **4**, 612–620.

124 S. F. Tan, S. W. Chee, G. Lin, M. Bosman, M. Lin, U. Mirsaidov and C. A. Nijhuis, Real-time imaging of the formation of Au-Ag core-shell nanoparticles, *J. Am. Chem. Soc.*, 2016, **138**, 5190–5193.

125 W. I. Liang, X. Zhang, Y. Zan, M. Pan, C. Czarnik, K. Bustillo, J. Xu, Y. H. Chu and H. Zheng, *In situ* study of Fe₃Pt-Fe₂O₃ core-shell nanoparticle formation, *J. Am. Chem. Soc.*, 2015, **137**, 14850–14853.

126 N. Ahmad, M. Bon, D. Passerone and R. Erni, Template-assisted *in situ* synthesis of Ag@Au bimetallic

nanostructures employing liquid-phase transmission electron microscopy, *ACS Nano*, 2019, **13**, 13333–13342.

127 M. Song, D. Zhang, D. Leng, J. Lee, Z. Yang, J. Chen, D. Li, L. Wang, G. Zhou, R. Yang and K. Zhou, *In situ* atomic observations of aggregation growth and evolution of penta-twinned gold nanocrystals, *Nat. Commun.*, 2024, **15**, 9217.

128 R. Mendoza-Cruz, J. P. Palomares-Báez, S. M. López-López, J. M. Montejano-Carrizales, J. L. Rodríguez López, M. José Yacamán and L. Bazán-Díaz, Experimental high-resolution observation of the truncated double-icosahedron structure: a stable twinned shell in alloyed Au-Ag core@shell nanoparticles, *Nano Lett.*, 2024, **24**, 4072–4081.

129 H. Kim, T. Y. Yoo, M. S. Bootharaju, J. H. Kim, D. Y. Chung and T. Hyeon, Noble metal-based multimetallic nanoparticles for electrocatalytic applications, *Adv. Sci.*, 2022, **9**, 2104054.

130 X. Geng, S. Li, J. Heo, Y. Peng, W. Hu, Y. Liu, J. Huang, Y. Ren, D. Li, L. Zhang and L. Luo, Grain-boundary-rich noble metal nanoparticle assemblies: synthesis, characterization, and reactivity, *Adv. Funct. Mater.*, 2022, **32**, 2204169.

131 X. Ma, F. Lin, X. Chen and C. Jin, Unveiling growth pathways of multiply twinned gold nanoparticles by *in situ* liquid cell transmission electron microscopy, *ACS Nano*, 2020, **14**, 9594–9604.

132 C. Zhu, S. Liang, E. Song, Y. Zhou, W. Wang, F. Shan, Y. Shi, C. Hao, K. Yin, T. Zhang, J. Liu, H. Zheng and L. Sun, *In situ* liquid cell transmission electron microscopy investigation on oriented attachment of gold nanoparticles, *Nat. Commun.*, 2018, **9**, 421.

133 Z. Q. Zhang, Y. C. Pei, M. J. Xiao, G. Hu, Z. P. Huang, T. Song, Q. Wang, W. Y. Huang, Y. Peng and H. L. Zhang, *In situ* observation of the crystal structure transition of Pt-Sn intermetallic nanoparticles during deactivation and regeneration, *Chem. Commun.*, 2021, **57**, 5454–5457.

134 M. Kamp, A. Tymoczko, U. Schürmann, J. Jakobi, C. Rehbock, K. Rätzke, S. Barcikowski and L. Kienle, Temperature-dependent ultrastructure transformation of Au-Fe nanoparticles investigated by *in situ* scanning transmission electron microscopy, *Cryst. Growth Des.*, 2018, **18**, 5434–5440.

135 T. Vystavel, G. Palasantzas, S. A. Koch, J. Th and M. De Hosson, Nanosized iron clusters investigated with *in situ* transmission electron microscopy, *Appl. Phys. Lett.*, 2003, **82**, 197–199.

136 K. Y. Jang, S. J. Ahn, J. H. Kwon, K. M. Nam and Y. H. Kim, Novel route from a wurtzite to a rock-salt structure in CoO nanocrystals: *in situ* transmission electron microscopy study, *J. Phys. Chem. C*, 2019, **123**, 10689–10694.

137 B. Jin, F. Zhang, G. Wu, T. Yuan, Q. Wang, H. Zhou, Y. Zhao, G. Zhang and X. Hong, Structural evolution induced by Au atom diffusion in Ag₂S, *Chem. Commun.*, 2019, **55**, 13176–13178.

138 A. Govender, E. J. Olivier, S. J. Haigh, D. Kelly, M. Smith, H. van Rensburg, R. P. Forbes and E. van Steen,

Performance of a NiFe₂O₄@Co core-shell fischer-tropsch catalyst: effect of low temperature reduction, *ACS Omega*, 2020, **5**, 32975–32983.

139 W. Xia, Y. Yang, Q. Meng, Z. Deng, M. Gong, J. Wang, D. Wang, Y. Zhu, L. Sun, F. Xu, J. Li and H. L. Xin, Bimetallic nanoparticle oxidation in three dimensions by chemically sensitive electron tomography and *in situ* transmission electron microscopy, *ACS Nano*, 2018, **12**, 7866–7874.

140 X. Shen, S. Dai, S. Zhang, Z. Lu, C. Zhang, G. W. Graham, Y. Lei, X. Pan and Z. Peng, Oxidation-induced atom diffusion and surface restructuring in faceted ternary Pt-Cu-Ni nanoparticles, *Chem. Mater.*, 2019, **31**, 1720–1728.

141 C. Ma, Y. Yun, T. Zhang, H. Suo, L. Yan, X. Shen, Y. Li and Y. Yang, Insight into the structural evolution of the cobalt oxides nanoparticles upon reduction process: an *in situ* transmission electron microscopy study, *ChemCatChem*, 2021, **13**, 4350–4354.

142 A. P. LaGrow, D. C. Lloyd, P. L. Gai and E. D. Boyes, *In situ* scanning transmission electron microscopy of Ni nanoparticle redispersion via the reduction of hollow NiO, *Chem. Mater.*, 2018, **30**, 197–203.

143 T. Wang, W. Lu, X. Xu, J. Qiu and S. F. Yu, Study of crystallization and coalescence of nanocrystals in amorphous glass at high temperature, *Inorg. Chem.*, 2019, **58**, 9500–9504.

144 D. Wang, Z. Zhao, B. Shi, J. X. Wang and J. F. Chen, Real-time imaging and quantitative evolution for pyrolysis of carbon dots-encapsulated metal-organic frameworks at the nanoscale by *in situ* environmental transmission electron microscopy, *ACS Appl. Mater. Interfaces*, 2023, **15**, 35358–35365.

145 T. S. Kim, J. Kim, H. C. Song, D. Kim, B. Jeong, J. Lee, J. W. Shin, R. Ryoo and J. Y. Park, Catalytic synergy on PtNi bimetal catalysts driven by interfacial intermediate structures, *ACS Catal.*, 2020, **10**, 10459–10467.

146 A. R. Poerwoprajitno, L. Gloag, J. Watt, S. Cheong, X. Tan, H. Lei, H. A. Tahini, A. Henson, B. Subhash, N. M. Bedford, B. K. Miller, P. B. O'Mara, T. M. Benedetti, D. L. Huber, W. Zhang, S. C. Smith, J. J. Gooding, W. Schuhmann and R. D. Tilley, A single-Pt-atom-on-Ru-nanoparticle electrocatalyst for CO-resilient methanol oxidation, *Nat. Catal.*, 2022, **5**, 231–237.

147 B. Song, Y. Yang, T. T. Yang, K. He, X. Hu, Y. Yuan, V. P. Dravid, M. R. Zachariah, W. A. Saidi, Y. Liu and R. Shahbazian-Yassar, Revealing high-temperature reduction dynamics of high-entropy alloy nanoparticles via *in situ* transmission electron microscopy, *Nano Lett.*, 2021, **21**, 1742–1748.

148 B. Song, Y. Yang, M. Rabbani, T. T. Yang, K. He, X. Hu, Y. Yuan, P. Ghildiyal, V. P. Dravid, M. R. Zachariah, W. A. Saidi, Y. Liu and R. Shahbazian-Yassar, *In situ* oxidation studies of high-entropy alloy nanoparticles, *ACS Nano*, 2020, **14**, 15131–15143.

149 E. Garnett, L. Mai and P. Yang, Introduction: 1D nanomaterials/nanowires, *Chem. Rev.*, 2019, **119**, 8955–8957.

150 I. Cho, J. Ko, D. D. O. Henriquez, D. Yang and I. Park, Recent advances in 1D nanostructure assembly and direct integration methods for device applications, *Small Methods*, 2024, **8**, 2400474.

151 Q. Wei, F. Xiong, S. Tan, L. Huang, E. H. Lan, B. Dunn and L. Mai, Porous one-dimensional nanomaterials: design, fabrication and applications in electrochemical energy storage, *Adv. Mater.*, 2017, **29**, 1602300.

152 T. Pham, K. Reidy, J. D. Thomsen, B. Wang, N. Deshmukh, M. A. Filler and F. M. Ross, Salt-assisted vapor-liquid-solid growth of 1D van der waals materials, *Adv. Mater.*, 2024, **36**, 2309360.

153 P. Sutter and E. Sutter, Tunable 1D van der waals nanostructures by vapor-liquid-solid growth, *Acc. Chem. Res.*, 2023, **56**, 3235–3245.

154 H. Wang, J. T. Wang, Z. X. Cao, W. J. Zhang, C. S. Lee, S. T. Lee and X. H. Zhang, A surface curvature oscillation model for vapour-liquid-solid growth of periodic one-dimensional nanostructures, *Nat. Commun.*, 2015, **6**, 6412.

155 Q. Sun, D. Pan, M. Li, J. Zhao, P. Chen, W. Lu and J. Zou, *In situ* TEM observation of the vapor-solid-solid growth of <001> InAs nanowires, *Nanoscale*, 2020, **12**, 11711–11717.

156 T. Hu, Y. Cao, S. M. Franzén, D. Jacobsson, M. S. Seifner, M. E. Messing and K. A. Dick, *In situ* manipulation of growth mechanisms in the vapor-solid-solid growth of GaP nanowires, *Adv. Mater. Interfaces*, 2024, 2400805.

157 E. Bellet-Amalric, F. Panciera, G. Patriarche, L. Travers, M. den Hertog, J. C. Harmand, F. Glas and J. Cibert, Regulated dynamics with two monolayer steps in vapor-solid-solid growth of nanowires, *ACS Nano*, 2022, **16**, 4397–4407.

158 S. Song, K. Y. Kim, S. H. Lee, K. K. Kim, K. Lee, W. Lee, H. Jeon and S. H. Ko, Recent advances in 1D nanomaterial-based bioelectronics for healthcare applications, *Adv. NanoBiomed Res.*, 2022, **2**, 2100111.

159 T. Jin, Q. Han, Y. Wang and L. Jiao, 1D nanomaterials: design, synthesis, and applications in sodium-ion batteries, *Small*, 2018, **14**, 1703086.

160 G. W. Liu, Y. Zhang, M. P. Thomas, A. Ullah, M. Pharr, B. S. Guiton and S. Banerjee, Negative thermal expansion HfV_2O_7 nanostructures for alleviation of thermal stress in nanocomposite coatings, *ACS Appl. Mater. Interfaces*, 2021, **13**, 44723–44732.

161 X. Li, S. Cheng, S. Deng, X. Wei, J. Zhu and Q. Chen, Direct observation of the layer-by-layer growth of ZnO nanopillar by *in situ* high resolution transmission electron microscopy, *Sci. Rep.*, 2017, **7**, 40911.

162 J. H. Kim, J. G. Kim, J. Song, T. S. Bae, K. H. Kim, Y. S. Lee, Y. Pang, K. H. Oh and H.-S. Chung, Investigation of the growth and *in situ* heating transmission electron microscopy analysis of Ag_2S-catalyzed ZnS nanowires, *Appl. Surf. Sci.*, 2018, **436**, 556–561.

163 Z. Fan, J. L. Maurice, I. Florea, W. Chen, L. Yu, S. Guilet, E. Cambril, X. Lafosse, L. Couraud, S. Bouchoule and P. Roca i Cabarrocas, *In situ* observation of droplet nanofluidics for yielding low-dimensional nanomaterials, *Appl. Surf. Sci.*, 2022, **573**, 151510.

164 S. Choi, J. Lee, M. Pin, J. H. Kwon, I. Kim, M. S. Yeom, C. S. Kim, H. S. Lee, S. J. Ahn, S. H. Yi and Y. H. Kim, Anisotropic atomistic evolution during the sublimation of polar InAs nanowires, *Nanoscale*, 2019, **11**, 6685–6692.

165 R. Boston, Z. Schnepp, Y. Nemoto, Y. Sakka and S. R. Hall, *In situ* TEM observation of a microcrucible mechanism of nanowire growth, *Science*, 2014, **344**, 623–626.

166 J. H. Chang, Y. T. Tseng, A. Y. Ho, H. Y. Lo, C. Y. Huang, S. C. Tsai, T. H. Yu, Y. L. Wu, H. K. Yen, P. H. Yeh, K. C. Lu and W. W. Wu, *In situ* TEM investigation of indium oxide/titanium oxide nanowire heterostructures growth through solid state reactions, *Mater. Charact.*, 2022, **187**, 111832.

167 K. M. Vailonis, K. Gnanasekaran, X. B. Powers, N. C. Gianneschi and D. M. Jenkins, Elucidating the growth of metal-organic nanotubes combining isoreticular synthesis with liquid-cell transmission electron microscopy, *J. Am. Chem. Soc.*, 2019, **141**, 10177–10182.

168 K. He, A. Nie, Y. Yuan, S. M. Ghodsi, B. Song, E. Firlar, J. Lu, Y. Lu, T. Shokuhfar, C. M. Megaridis and R. Shahbazian-Yassar, *In situ* transmission electron microscopy explores a new nanoscale pathway for direct gypsum formation in aqueous solution, *ACS Appl. Nano Mater.*, 2018, **1**, 5430–5440.

169 K. Gnanasekaran, K. M. Vailonis, D. M. Jenkins and N. C. Gianneschi, *In situ* monitoring of the seeding and growth of silver metal-organic nanotubes by liquid-cell transmission electron microscopy, *ACS Nano*, 2020, **14**, 8735–8743.

170 É. Ngo, W. Wang, P. Bulkin, I. Florea, M. Foldyna, P. Roca i Cabarrocas and J. L. Maurice, Liquid-assisted vapor-solid-solid silicon nanowire growth mechanism revealed by *in situ* TEM when using Cu-Sn bimetallic catalysts, *J. Phys. Chem. C*, 2021, **125**, 19773–19779.

171 K. Gnanasekaran, J. Korpanty, O. Berger, N. Hampu, M. Halperin-Sternfeld, D. Cohen-Gerassi, L. Adler-Abramovich and N. C. Gianneschi, Dipeptide nanostructure assembly and dynamics via *in situ* liquid-phase electron microscopy, *ACS Nano*, 2021, **15**, 16542–16551.

172 X. Ye, M. R. Jones, L. B. Frechette, Q. Chen, A. S. Powers, P. Ercius, G. Dunn, G. M. Rotskoff, S. C. Nguyen, V. P. Adiga, A. Zettl, E. Rabani, P. L. Geissler and A. P. Alivisatos, Single-particle mapping of nonequilibrium nanocrystal transformations, *Science*, 2016, **354**, 874–877.

173 K. Aliyah, J. Lyu, C. Goldmann, T. Bizien, C. Hamon, D. Alloyeau and D. Constantin, real-time *in situ* observations reveal a double role for ascorbic acid in the anisotropic growth of silver on gold, *J. Phys. Chem. Lett.*, 2020, **11**, 2830–2837.

174 M. Dong, W. Wang, W. Wei, X. Hu, M. Qin, Q. Zhang, L. Sun and F. Xu, Understanding the ensemble of growth behaviors of sub-10-nm silver nanorods using *in situ* liquid cell transmission electron microscopy, *J. Phys. Chem. C*, 2019, **123**, 21257–21264.

175 M. S. A. Asghar, B. J. Inkson and G. Möbus, *In situ* formation of 1D nanostructures from ceria nanoparticle dispersions by liquid cell TEM irradiation, *J. Mater. Sci.*, 2020, **55**, 2815–2825.

176 H. G. Liao, L. Cui, S. Whitelam and H. Zheng, Real-time imaging of Pt$_3$Fe nanorod growth in solution, *Science*, 2012, **336**, 1011–1014.

177 K. Y. Niu, M. Liu, K. A. Persson, Y. Han and H. Zheng, Strain-mediated interfacial dynamics during Au-PbS core-shell nanostructure formation, *ACS Nano*, 2016, **10**, 6235–6240.

178 M. Xin, Z. Chang, H. Luo, R. Zeng, G. Zhang and Q. Lei, An electrical super-insulator prototype of 1D gas-solid Al$_2$O$_3$ nanocell, *Nano Energy*, 2017, **39**, 95–100.

179 Y. Ma, W. Gao, H. Shan, W. Chen, W. Shang, P. Tao, C. Song, C. Addiego, T. Deng, X. Pan and J. Wu, Platinum-based nanowires as active catalysts toward oxygen reduction reaction: *in situ* observation of surface-diffusion-assisted, solid-state oriented attachment, *Adv. Mater.*, 2017, **29**, 1703460.

180 L. C. Campos, M. Tonezzer, A. S. Ferlauto, V. Grillo, R. Magalhães-Paniago, S. Oliveira, L. O. Ladeira and R. G. Lacerda, Vapor-solid-solid growth mechanism driven by epitaxial match between solid AuZn alloy catalyst particles and ZnO nanowires at low temperatures, *Adv. Mater.*, 2008, **20**, 1499–1504.

181 F. Panciera, Z. Baraissov, G. Patriarche, V. G. Dubrovskii, F. Glas, L. Travers, U. Mirsaidov and J. C. Harmand, Phase selection in self-catalyzed GaAs nanowires, *Nano Lett.*, 2020, **20**, 1669–1675.

182 M. Marnauza, R. Sjökvist, S. Lehmann and K. A. Dick, Diameter control of GaSb nanowires revealed by *in situ* environmental transmission electron microscopy, *J. Phys. Chem. Lett.*, 2023, **14**, 7404–7410.

183 B. M. Hudak, Y. J. Chang, L. Yu, G. Li, D. N. Edwards, M. E. Park and B. S. Guiton, Understanding nanomaterial synthesis with *in situ* transmission electron microscopy, *Microsc. Microanal.*, 2015, **21**, 1507–1508.

184 W. Xin, I. M. D. Rosa, Y. Cao, X. Yin, H. Yu, P. Ye, L. Carlson and J. M. Yang, Ultrasonication-assisted synthesis of high aspect ratio gold nanowires on a graphene template and investigation of their growth mechanism, *Chem. Commun.*, 2018, **54**, 4124–4127.

185 Y. C. Chou, F. Panciera, M. C. Reuter, E. A. Stach and F. M. Ross, Nanowire growth kinetics in aberration corrected environmental transmission electron microscopy, *Chem. Commun.*, 2016, **52**, 5686–5689.

186 X. Zhao, S. Sun, F. Yang and Y. Li, Atomic-scale evidence of catalyst evolution for the structure controlled growth of single-walled carbon nanotubes, *Acc. Chem. Res.*, 2022, **55**, 3334–3344.

187 L. Zhang, M. He, T. W. Hansen, J. Kling, H. Jiang, E. I. Kauppinen, A. Loiseau and J. B. Wagner, Growth termination and multiple nucleation of single-wall carbon nanotubes evidenced by *in situ* transmission electron microscopy, *ACS Nano*, 2017, **11**, 4483–4493.

188 R. Ma, L. Qiu, L. Zhang, D. M. Tang, Y. Wang, B. Zhang, F. Ding, C. Liu and H. M. Cheng, Nucleation of single-wall carbon nanotubes from faceted Pt catalyst particles revealed by *in situ* transmission electron microscopy, *ACS Nano*, 2022, **16**, 16574–16583.

189 X. Huang, R. Farra, R. Schlögl and M. G. Willinger, Growth and termination dynamics of multiwalled carbon nanotubes at near Ambient pressure: an *in situ* transmission electron microscopy study, *Nano Lett.*, 2019, **19**, 5380–5387.

190 H. Fan, L. Qiu, A. Fedorov, M. G. Willinger, F. Ding and X. Huang, Dynamic state and active structure of Ni-Co catalyst in carbon nanofiber growth revealed by *in situ* transmission electron microscopy, *ACS Nano*, 2021, **15**, 17895–17906.

191 M. Choe, H. Chung, W. Kim, Y. Jang, Z. Wang and Z. Lee, *In situ* transmission electron microscopy of temperature-dependent carbon nanofiber and carbon nanotube growth from ethanol vapor, *Carbon*, 2024, **219**, 118843.

192 H. Yoshida, S. Takeda, T. Uchiyama, H. Kohno and Y. Homma, Atomic-scale *in situ* observation of carbon nanotube growth from solid state iron carbide nanoparticles, *Nano Lett.*, 2008, **8**, 2082–2086.

193 H. Zhang, H. M. Cheng and P. Ye, 2D nanomaterials: beyond graphene and transition metal dichalcogenides, *Chem. Soc. Rev.*, 2018, **47**, 6009–6012.

194 H. Zhang, Ultrathin two-dimensional nanomaterials, *ACS Nano*, 2015, **9**, 9451–9469.

195 Y. Zhu, L. Peng, Z. Fang, C. Yan, X. Zhang and G. Yu, Structural engineering of 2D nanomaterials for energy storage and catalysis, *Adv. Mater.*, 2018, **30**, 1706347.

196 Y. Wu, Y. Wu, Y. Sun, W. Zhao and L. Wang, 2D nanomaterials reinforced organic coatings for marine corrosion protection: state of the art, challenges, and future prospectives, *Adv. Mater.*, 2024, **36**, 2312460.

197 Y. Chao, Y. Han, Z. Chen, D. Chu, Q. Xu, G. Wallace and C. Wang, Multiscale structural design of 2D nanomaterials-based flexible electrodes for wearable energy storage applications, *Adv. Sci.*, 2024, **11**, 2305558.

198 C. Tan, X. Cao, X. J. Wu, Q. He, J. Yang, X. Zhang, J. Chen, W. Zhao, S. Han, G. H. Nam, M. Sindoro and H. Zhang, Recent advances in ultrathin two-dimensional nanomaterials, *Chem. Rev.*, 2017, **117**, 6225–6331.

199 X. Li, M. Sun, C. Shan, Q. Chen and X. Wei, Mechanical properties of 2D materials studied by *in situ* microscopy techniques, *Adv. Mater. Interfaces*, 2018, **5**, 1701246.

200 W. Sohn, M. Kim and H. W. Jang, Atomic-scale insights into the 2D materials from aberration-corrected scanning transmission electron microscopy: progress and future, *Small Sci.*, 2024, **4**, 2300073.

201 R. Luo, M. Gao, C. Wang, J. Zhu, R. Guzman and W. Zhou, Probing functional structures, defects, and interfaces of 2D transition metal dichalcogenides by electron microscopy, *Adv. Funct. Mater.*, 2024, **34**, 2307625.

202 Y. Zhang, Z. Zhang, Y. Cheng, F. Cheng, L. Wang, N. Liu, L. Li, J. Su and Y. Gao, *In situ* TEM observation of controlled growth of two-dimensional WS$_2$ with vertically

aligned layers and high-temperature stability, *Nano Energy*, 2020, **67**, 104221.

203 N. Kondekar, M. G. Boebinger, M. Tian, M. H. Kirmani and M. T. McDowell, The effect of nickel on MoS_2 growth revealed with *in situ* transmission electron microscopy, *ACS Nano*, 2019, **13**, 7117–7126.

204 X. Sang, X. Li, A. A. Puretzky, D. B. Geohegan, K. Xiao and R. R. Unocic, Atomic insight into thermolysis-driven growth of 2D MoS_2, *Adv. Funct. Mater.*, 2019, **29**, 1902149.

205 C. N. S. Kumar, M. Konrad, V. S. K. Chakravadhanula, S. Dehm, D. Wang, W. Wenzel, R. Krupke and C. Kübel, Nanocrystalline graphene at high temperatures: insight into nanoscale processes, *Nanoscale Adv.*, 2019, **1**, 2485–2494.

206 D. S. Gavhane, A. D. Sontakke and M. A. van Huis, Selective vertical and horizontal growth of 2D WS_2 revealed by *in situ* thermolysis using transmission electron microscopy, *Adv. Funct. Mater.*, 2022, **32**, 2106450.

207 M. Wang, J. H. Kim, S. S. Han, M. Je, J. Gil, C. Noh, T. J. Ko, K. S. Lee, D. I. Son, T. S. Bae, H. I. Ryu, K. H. Oh, Y. Jung, H. Choi, H. S. Chung and Y. Jung, Structural evolutions of vertically aligned two-dimensional MoS_2 layers revealed by *in situ* heating transmission electron microscopy, *J. Phys. Chem. C*, 2019, **123**, 27843–27853.

208 B. C. Bayer, R. Kaindl, M. Reza Ahmadpour Monazam, T. Susi, J. Kotakoski, T. Gupta, D. Eder, W. Waldhauser and J. C. Meyer, Atomic-scale *in situ* observations of crystallization and restructuring processes in two-dimensional MoS_2 films, *ACS Nano*, 2018, **12**, 8758–8769.

209 H. Inani, D. H. Shin, J. Madsen, H. Jeong, M. H. Kwon, N. McEvoy, T. Susi, C. Mangler, S. W. Lee, K. Mustonen and J. Kotakoski, Step-by-step atomic insights into structural reordering from 2D to 3D MoS_2, *Adv. Funct. Mater.*, 2021, **31**, 2008395.

210 J. Y. Zhang, L. P. Xiao and M. Lu, *In situ* liquid-phase transmission electron microscopy for two-dimensional energy materials, *Sci. China: Chem.*, 2025, **68**, 414–429.

211 G. Zhu, Y. Jiang, F. Lin, H. Zhang, C. Jin, J. Yuan, D. Yang and Z. Zhang, *In situ* study of the growth of two-dimensional palladium dendritic nanostructures using liquid-cell electron microscopy, *Chem. Commun.*, 2014, **50**, 9447–9450.

212 J. H. Park, N. M. Schneider, J. M. Grogan, M. C. Reuter, H. H. Bau, S. Kodambaka and F. M. Ross, Control of electron beam-induced Au nanocrystal growth kinetics through solution chemistry, *Nano Lett.*, 2015, **15**, 5314–5320.

213 B. Jin, H. Wang, M. L. Sushko, C. Jin and R. Tang, The formation and shape transformation mechanism of a triangular Au nanoplate revealed by liquid-cell TEM, *Nanoscale*, 2020, **12**, 19592–19596.

214 J. Zhang, Y. Jiang, Q. Fan, M. Qu, N. He, J. Deng, Y. Sun, J. Cheng, H. G. Liao and S. G. Sun, Atomic scale tracking of single layer oxide formation: self-peeling and phase transition in solution, *Small Methods*, 2021, **5**, 2001234.

215 J. Yang, Z. Zeng, J. Kang, S. Betzler, C. Czarnik, X. Zhang, C. Ophus, C. Yu, K. Bustillo, M. Pan, J. Qiu, L. W. Wang

and H. Zheng, Formation of two-dimensional transition metal oxide nanosheets with nanoparticles as intermediates, *Nat. Mater.*, 2019, **18**, 970–976.

216 K. S. Vasu, E. Prestat, J. Abraham, J. Dix, R. J. Kashtiban, J. Beheshtian, J. Sloan, P. Carbone, M. Neek-Amal, S. J. Haigh, A. K. Geim and R. R. Nair, Van der waals pressure and its effect on trapped interlayer molecules, *Nat. Commun.*, 2016, **7**, 12168.

217 Q. Zhang, X. Peng, Y. Nie, Q. Zheng, J. Shangguan, C. Zhu, K. C. Bustillo, P. Ercius, L. Wang, D. T. Limmer and H. Zheng, Defect-mediated ripening of core-shell nanostructures, *Nat. Commun.*, 2022, **13**, 2211.

218 J. Zhang, G. Li, H. G. Liao and S. G. Sun, Tracking the atomic pathways of Pt_3Ni-$Ni(OH)_2$ core-shell structures at the gas-liquid interface by *in situ* liquid cell TEM, *Sci. China Chem.*, 2020, **63**, 513–518.

219 J. Zhang, Atomic-scale imaging of the growth and transformation of Pt_3Ni-NiO nanoparticles, *New J. Chem.*, 2021, **45**, 2217–2220.

220 L. Xiao, G. Wang, X. Huang, S. Zhou, R. Zhou, Y. Jiang, S. Liu, G. Li, H. Zheng, S. G. Sun and H. G. Liao, Efficient CO_2 reduction MOFs derivatives transformation mechanism revealed by *in situ* liquid phase TEM, *Appl. Catal. B Environ.*, 2022, **307**, 121164.

221 W. Wang, H. Yan, U. Anand and U. Mirsaidov, Visualizing the conversion of metal-organic framework nanoparticles into hollow layered double hydroxide nanocages, *J. Am. Chem. Soc.*, 2021, **143**, 1854–1862.

222 X. Peng, P. M. Pelz, Q. Zhang, P. Chen, L. Cao, Y. Zhang, H. G. Liao, H. Zheng, C. Wang, S. G. Sun and M. C. Scott, Observation of formation and local structures of metal-organic layers via complementary electron microscopy techniques, *Nat. Commun.*, 2022, **13**, 5197.

223 N. A. Jose, J. J. Varghese, S. H. Mushrif, H. C. Zeng and A. A. Lapkin, Assembly of two-dimensional metal organic framework superstructures via solvent-mediated oriented attachment, *J. Phys. Chem. C*, 2021, **125**, 22837–22847.

224 Z. Ou, Z. Wang, B. Luo, E. Luijten and Q. Chen, Kinetic pathways of crystallization at the nanoscale, *Nat. Mater.*, 2020, **19**, 450–455.

225 W. Wang, S. W. Chee, H. Yan, I. Erofeev and U. Mirsaidov, Growth dynamics of vertical and lateral layered double hydroxide nanosheets during electrodeposition, *Nano Lett.*, 2021, **21**, 5977–5983.

226 B. K. Choi, J. Kim, Z. Luo, J. Kim, J. H. Kim, T. Hyeon, S. Mehraeen, S. Park and J. Park, Shape transformation mechanism of gold nanoplates, *ACS Nano*, 2023, **17**, 2007–2018.

227 D. Alloyeau, W. Dachraoui, Y. Javed, H. Belkahla, G. Wang, H. Lecoq, S. Ammar, O. Ersen, A. Wisnet, F. Gazeau and C. Ricolleau, Unravelling kinetic and thermodynamic effects on the growth of gold nanoplates by liquid transmission electron microscopy, *Nano Lett.*, 2015, **15**, 2574–2581.

228 M. Dong, R. Fu, H. Min, Q. Zhang, H. Dong, Y. Pan, L. Sun, W. Wei, M. Qin, Z. Zhu and F. Xu, *In situ* liquid cell transmission electron microscopy investigation on the

dissolution-regrowth mechanism dominating the shape evolution of silver nanoplates, *Cryst. Growth Des.*, 2021, **21**, 1314–1322.

229 J. Yu, X. Y. Li, J. Miao, W. Yuan, S. Zhou, B. Zhu, Y. Gao, H. Yang, Z. Zhang and Y. Wang, Atomic mechanism in layer-by-layer growth via surface reconstruction, *Nano Lett.*, 2019, **19**, 4205–4210.

230 Y. Liu, L. Xu, L. Zhang, Z. Dong, S. Wang and L. Luo, Direct visualization of atomic-scale graphene growth on Cu through environmental transmission electron microscopy, *ACS Appl. Mater. Interfaces*, 2020, **12**, 52201–52207.

231 J. Kling, T. W. Hansen and J. B. Wagner, Quantifying the growth of individual graphene layers by *in situ* environmental transmission electron microscopy, *Carbon*, 2016, **99**, 261–266.

232 K. Yu, W. Zhao, X. Wu, J. Zhuang, X. Hu, Q. Zhang, J. Sun, T. Xu, Y. Chai, F. Ding and L. Sun, *In situ* atomic-scale observation of monolayer graphene growth from SiC, *Nano Res.*, 2018, **11**, 2809–2820.

233 N. Kurtyka, B. van Devener, B. W. Chung and L. W. I. McDonald, *In situ* liquid cell transmission electron microscopy study of studtite particle formation and growth via electron beam radiolysis, *ACS Omega*, 2023, **8**, 48336–48343.

234 K. L. Tai, C. W. Huang, R. F. Cai, G. M. Huang, Y. T. Tseng, J. Chen and W. W. Wu, Atomic-scale fabrication of in-plane heterojunctions of few-layer MoS$_2$ via *in situ* scanning transmission electron microscopy, *Small*, 2020, **16**, 1905516.

235 Y. Fei, T. Tong, J. Bao and Y. H. Hu, *In situ* observation of electron-beam-induced NaH decomposition in graphene nanoreactors by transmission electron microscopy, *J. Phys. Chem. Lett.*, 2023, **14**, 1–8.

236 T. Clark, C. A. Taylor, C. M. Barr and K. Hattar, Sample preparation and experimental design for *in situ* multi-beam transmission electron microscopy irradiation experiments, *J. Vis. Exp.*, 2022, e61293.

237 H. Ma, D. Kim, S. I. Park, B. K. Choi, G. Park, H. Baek, H. Lee, H. Kim, J. S. Yu, W. C. Lee, J. Park and J. Yang, Direct observation of off-stoichiometry-induced phase transformation of 2D CdSe quantum nanosheets, *Adv. Sci.*, 2023, **10**, 2205690.

238 K. K. Neelisetty, X. Mu, S. Gutsch, A. Vahl, A. Molinari, F. Von Seggern, M. Hansen, T. Scherer, M. Zacharias, L. Kienle, V. K. Chakravadhanula and C. Kübel, Electron beam effects on oxide thin films—structure and electrical property correlations, *Microsc. Microanal.*, 2019, **25**, 592–600.

239 O. Dyck, A. R. Lupini, P. D. Rack, J. Fowlkes and S. Jesse, Controlling hydrocarbon transport and electron beam induced deposition on single layer graphene: Toward atomic scale synthesis in the scanning transmission electron microscope, *Nano Sel.*, 2022, **3**, 643–654.

240 S. J. Kim, K. S. Dae, J. Y. Park, J. Y. Lee and J. M. Yuk, Hollow Ag$_2$S nanosphere formation via electron beam-assisted oxidative etching of Ag nanoparticles, *Chem. Commun.*, 2017, **53**, 11122–11125.

241 Y. Ding, K. C. Pradel and Z. L. Wang, *In situ* transmission electron microscopy observation of ZnO polar and non-polar surfaces structure evolution under electron beam irradiation, *J. Appl. Phys.*, 2016, **119**, 015305.

242 Y. T. Tseng, L. S. Lu, F. C. Shen, C. H. Wang, H. Y. Sung, W. H. Chang and W. W. Wu, *In situ* atomic-scale observation of monolayer MoS$_2$ devices under high-voltage biasing via transmission electron microscopy, *Small*, 2022, **18**, 2106411.

243 Y. T. Tseng, K. L. Tai, C. W. Huang, C. Y. Huang and W. W. Wu, Atomic-scale localized thinning and reconstruction of two-dimensional WS$_2$ layers through *in situ* transmission electron microscopy/scanning transmission electron microscopy, *J. Phys. Chem. C*, 2020, **124**, 14935–14940.

244 S. Dai, M. He and J. Zhu, E-beam-induced *in situ* structural transformation in one-dimensional nanomaterials, *Sci. Bull.*, 2015, **60**, 71–75.

245 M. Owusu-Mensah, J. Cooper, A. L. Morales, K. Yano, S. D. Taylor, D. K. Schreiber, B. P. Uberuaga and D. Kaoumi, Surprisingly high irradiation-induced defect mobility in Fe$_3$O$_4$ as revealed through *in situ* transmission electron microscopy, *Mater. Charact.*, 2022, **187**, 111863.

246 S. Y. Kim, J. H. Kim, T. Jeong, K. B. Kim, H. J. Kim, K. M. Nam, S. J. Ahn, J. H. Kwon and Y. H. Kim, Accelerated decomposition of Bi$_2$S$_3$ nanorods in water under an electron beam: a liquid phase transmission electron microscopy study, *Nanotechnology*, 2021, **32**, 195702.

247 H. Zhang, W. Wang, T. Xu, F. Xu and L. Sun, Phase transformation at controlled locations in nanowires by *in situ* electron irradiation, *Nano Res.*, 2020, **13**, 1912–1919.

248 X. Huang, T. Jones, H. Fan and M. G. Willinger, Atomic-scale observation of irradiation-induced surface oxidation by *in situ* transmission electron microscopy, *Adv. Mater. Interfaces*, 2016, **3**, 1600751.

249 Z. Dang, J. Shamsi, F. Palazon, M. Imran, Q. A. Akkerman, S. Park, G. Bertoni, M. Prato, R. Brescia and L. Manna, *In situ* transmission electron microscopy study of electron beam-induced transformations in colloidal cesium lead halide perovskite nanocrystals, *ACS Nano*, 2017, **11**, 2124–2132.

250 I. G. Gonzalez-Martinez, A. Bachmatiuk, V. Bezugly, J. Kunstmann, T. Gemming, Z. Liu, G. Cuniberti and M. H. Rümmeli, Electron-beam induced synthesis of nanostructures: a review, *Nanoscale*, 2016, **8**, 11340–11362.

251 X. Guo, H. Xu, Y. Tang, Z. Yang, F. Dou, W. Li, Q. Li and H. Pang, Confining iodine into metal-organic framework derived metal-nitrogen-carbon for long-life aqueous zinc-iodine batteries, *Adv. Mater.*, 2024, **36**, 2408317.

252 C. Guo, Y. Cao, Y. Gao, C. Zhi, Y. X. Wang, Y. Luo, X. J. Yang and X. Luo, Cobalt single-atom electrocatalysts enhanced by hydrogen-bonded organic frameworks for long-lasting zinc-iodine batteries, *Adv. Funct. Mater.*, 2024, **34**, 2314189.

253 E. M. You, Y. Gu, J. Yi, D. Y. Wu, J. F. Li and Z. Q. Tian, Unraveling the multilayer solid-electrolyte interphase in lithium batteries through depth-sensitive plasmon-enhanced Raman spectroscopy: a theoretical and experimental study, *Electrochim. Acta*, 2024, **498**, 144689.

254 J. Wang, W. Tang, Z. Zhu, Y. Lin, L. Zhao, H. Chen, X. Qi, X. Niu, R. Wu and J. S. Chen, Stabilizing lattice oxygen of Bi_2O_3 by interstitial insertion of indium for efficient formic acid electrosynthesis, *Angew. Chem., Int. Ed.*, 2025, e202423658.

255 A. Duan, S. Luo, Y. Tang, Y. Feng, M. Li, B. Zhang and W. Sun, *In situ* monitoring of dynamic adsorption-induced interfacial buffering toward highly stable zinc metal batteries, *Adv. Energy Mater.*, 2025, 2404693.

256 É. A. Santos, M. C. Policano, M. J. Pinzón, I. Galantini, V. A. Gonçalves, F. C. B. Maia, L. J. A. Macedo, G. Doubek, R. G. Freitas and H. Zanin, Operando FTIR study on water additive in lithium-sulfur batteries to mitigate shuttle effect, *J. Energy Chem.*, 2024, **98**, 702–713.

257 X. Liu, M. Zhang, X. Wang, Y. Peng, Y. Liu, S. Ullah, Z. Duan, W. Gao, B. Song, M. Wei, J. He, Z. Li and Y. Wu, Evidence of quasi-Na metallic clusters in sodium ion batteries through *in situ* X-Ray diffraction, *Adv. Mater.*, 2025, **37**, 2410673.

258 X. Li, W. Hao, H. Wang, T. Li, D. Trikkaliotis, X. Zhou, D. Hou, K. Chang, A. M. Hashem, Y. Liu, Z. Yang, S. Cao, G. Hwang, G. Z. Kyzas, S. Yang, C. B. Mullins, C. M. Julien and L. Zhu, Lithium storage mechanisms and electrochemical behavior of a molybdenum disulfide nanoparticle anode, *Energy Environ. Mater.*, 2024, e12855.

259 L. Li, A. Huang, H. Jiang, Y. Li, X. Pan, T. Y. Chen, H. Y. Chen and S. Peng, Encapsulation of Sn sub-nanoclusters in multichannel carbon matrix for high-performance potassium-ion batteries, *Angew. Chem., Int. Ed.*, 2024, **63**, e202412077.

260 K. Romanenko, N. Avdievich and E. Foy, A universal plug-and-play approach to *in situ* multinuclear magnetic resonance analysis of electrochemical phenomena in commercial battery cells, *J. Am. Chem. Soc.*, 2024, **146**, 29407–29416.

261 M. Gabrijelčič, B. Tratnik, G. Kapun, E. Tchernychova, N. Z. Logar, A. Krajnc, R. Dominko and A. Vizintin, Probing sodium structures and dynamics in hard carbon for Na-ion batteries using ^{23}Na operando solid-state NMR spectroscopy, *J. Mater. Chem. A*, 2025, **13**, 1042–1056.

262 I. E. Gunathilaka, S. A. Ferdousi, F. Chen, M. Armand, A. A. H. Padua, P. C. Howlett, M. Forsyth and L. A. O'Dell, Studying the growth and morphology of metal microstructures in sodium metal batteries with ionic liquid electrolytes by operando ^{23}Na NMR spectroscopy, *Nano Energy*, 2025, **133**, 110479.

263 Y. Kwon, A. Svirinovsky-Arbeli, J. C. Hestenes, P. J. Buitrago Botero, K. R. M. Corpus, P. Lepucki, O. Pecher and L. E. Marbella, Elucidating the role of cathode identity: voltage-dependent reversibility of anode-free batteries, *Chem*, 2024, **10**, 3159–3183.

264 D. J. Yun, S. Kim, S. Heo, H. Choi, J. Baik, J. Chung, S. Park, D. Yu, J. Lee, S. Kang, C. Jung and D. S. Ko, Understanding the role of metal Interlayers in all-solid-electrolyte batteries using operando X-Ray photoelectron spectroscopy, *Adv. Energy Mater.*, 2024, **14**, 2470140.

265 R. Lee, T. S. Nunney, M. Isaacs, R. G. Palgrave and A. Dey, Monitoring the behavior of Na ions and solid electrolyte interphase formation at an aluminum/ionic liquid electrode/electrolyte interface via operando electrochemical X-ray photoelectron spectroscopy, *ACS Appl. Mater. Interfaces*, 2024, **16**, 35675–35685.

266 F. G. Capone, J. Sottmann, V. Meunier, L. P. Ramírez, A. Grimaud, A. Iadecola, M. Scardamaglia, J. P. Rueff and R. Dedryvère, Operando observation of the dynamic SEI formation on a carbonaceous electrode by near-ambient pressure XPS, *Energy Environ. Sci.*, 2024, **17**, 1509–1519.

267 A. Krishnan, D. C. Lee, I. Slagle, S. Ahsan, S. Mitra, E. Read and F. M. Alamgir, Monitoring redox processes in lithium-ion batteries by laboratory-scale operando X-ray emission spectroscopy, *ACS Appl. Mater. Interfaces*, 2024, **16**, 16096–16105.

268 W. Huang, M. Cao, H. Mao, L. An, Z. Chen, W. Xu, X. Li and H. Wei, Temperature-tunable operando nondestructive detection of electronic and geometrical structures in battery electrodes, *Anal. Chem.*, 2024, **96**, 1178–1184.

269 Z. Li, H. Yang, W. Cheng and L. Tian, Recent progress of *in situ*/operando characterization techniques for electrocatalytic energy conversion reaction, *Chin. Chem. Lett.*, 2024, **35**, 109237.