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Platinum–copper nanowire networks with enhanced CO tolerance toward methanol oxidation electrocatalysis†

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Developing platinum-based electrocatalysts with high CO tolerance for the methanol oxidation reaction (MOR) is crucial for the practical application of direct methanol fuel cells (DMFCs). Herein, we employed a straightforward one-step method to synthesize Pt_xCu_y network nanowires (NWNs), which exhibit the advantages of structural stability and bimetallic ensembles. The synergistic effect of compressive strain and the ligand effect, induced by Cu incorporation, can effectively lower the d-band center of Pt, thereby weakening the adsorption strength of CO on the catalyst surface. The optimized Pt₄₂Cu₅₈ NWNs deliver a peak mass activity of 1.33 A mg_{Pt}^{−1} and a specific activity of 4.43 mA cm^{−2} for the MOR, which are 3.03 and 4.03 times higher than those of commercial Pt/C, respectively. The CO stripping and *in situ* Fourier transform infrared spectroscopy results indicate their high anti-CO poisoning ability and methanol activation capacity. Moreover, the Pt₄₂Cu₅₈ NWNs also exhibit an excellent stability with high current densities observed after 3600 s of operation due to the enhanced CO tolerance and the stable three-dimensional (3D) network structure. This work provides a feasible strategy to suppress CO poisoning during the MOR and obtain highly efficient anode catalysts with enhanced durability in the DMFC field.

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1 Introduction

Direct methanol fuel cells (DMFCs) represent a promising electrochemical power source, particularly in view of the usage of liquid methanol, which offers advantages in terms of convenience and security for transportation and storage through the existing gasoline infrastructure, as compared to hydrogen.¹ Nevertheless, the sluggish kinetics of the methanol oxidation reaction (MOR), which entails six electron transfer steps at the anode, hinders the progress of DMFCs.^{2,3} During the stepwise dehydrogenation of methanol, carbonyl intermediates, particularly CO, tend to strongly adsorb onto the catalyst surface, blocking active sites for follow-up reactions.^{4,5} Platinum (Pt) nanoparticle electrocatalysts are presently recognized as the most effective anode catalysts.^{1,6,7} However, their catalytic efficiency is significantly hindered by the poisoning effect of the CO intermediate and Ostwald ripening during the MOR process.^{5,8–10}

The introduction of late transition metals (LTMs, such as Fe,¹¹ Co,¹² Ni¹³ and Cu¹⁴) into active Pt sites has emerged as a significant strategy to enhance the accelerated removal of CO. Incorporating LTMs enables the modification of the electronic structure of Pt and induces compressive strain on its surface, which results from the relatively smaller lattice constants of LTMs compared to those of Pt. Both the strain and ligand effects generated in the alloy can finely tune the d-band center, thereby reducing the adsorption strength of CO* intermediates on Pt active sites.^{15–20} Moreover, LTMs are abundant in nature and cost-effective, which can significantly reduce the expense of platinum-based catalysts, making them a more viable option for the MOR. Ultrathin one-dimensional nanowires (1D NWs), characterized by a typical diameter of less than 5 nm, exhibit distinctive properties such as high aspect ratios and enhanced atomic utilization efficiency.²¹ The NWs can readily self-assemble into a three-dimensional (3D) network structure, characterized by a large surface area with abundant exposed active sites, enhanced conductivity, and a stable structure that resists Ostwald ripening and degradation during the reaction.²² Therefore, adjusting the d-band center and enhancing structural stability are crucial for designing high-performance catalysts with superior anti-CO poisoning capabilities.

By comparing with Fe,²³ Ni,²⁴ Co,²⁵ etc., the Cu incorporation can optimize the d-band center to achieve an optimal balance between reduced CO binding energy and enhanced OH

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adsorption.^{26,27} The integration of Cu doping strategies with the distinctive structural advantages of NWs offers a promising approach for the design of MOR catalysts. Significant advancements have been achieved in PtCu NW systems. For instance, Zhang *et al.* synthesized screw thread-like Pt–Cu NWs featuring high-index facets, which provided an abundance of catalytically active sites and consequently enhanced the intrinsic catalytic activity.²⁸ Other researches also demonstrated that the construction of atomic steps²⁹ and the controlled introduction of surface defects^{26,29,30} can effectively increase active site density on NW surfaces, thereby boosting MOR performance. However, the critical role of composition-dependent strain–ligand synergy in optimizing CO tolerance and enhancing MOR activity remains underestimated. A more thorough investigation of this regulatory mechanism could reveal new avenues for the design of high-performance MOR catalysts.

Herein, we employed a simple one-step method to introduce Cu into Pt active sites, forming Pt_xCu_y NW networks (NWNs) for

the MOR. The characterization and *in situ* experimental results demonstrate that the synergistic effects of compressive strain and ligand effects, induced by Cu incorporation, can reduce CO adsorption, enhance CO oxidation, and liberate Pt active sites. This promotes MOR activity and improves resistance to CO poisoning. The optimized Pt₄₂Cu₅₈ NWNs demonstrate a peak mass activity (MA) of 1.33 A mg_{Pt}^{−1} and specific activity (SA) of 4.43 mA cm^{−2} for the MOR, 4.03 times and 5.03 times that of commercial Pt/C, respectively. Moreover, the Pt₄₂Cu₅₈ NWNs exhibit excellent stability with high current densities observed after 3600 s of continuous operation attributed to their superior resistance to CO poisoning and the stable 3D NWN structure.

2 Results and discussion

2.1 Synthesis and characterization

The Pt_xCu_y alloy NWNs were synthesized *via* a simple one-step method, as illustrated in Fig. 1a, using H₂PtCl₆ and CuCl₂·2H₂O as metal precursors, sodium borohydride as a strong

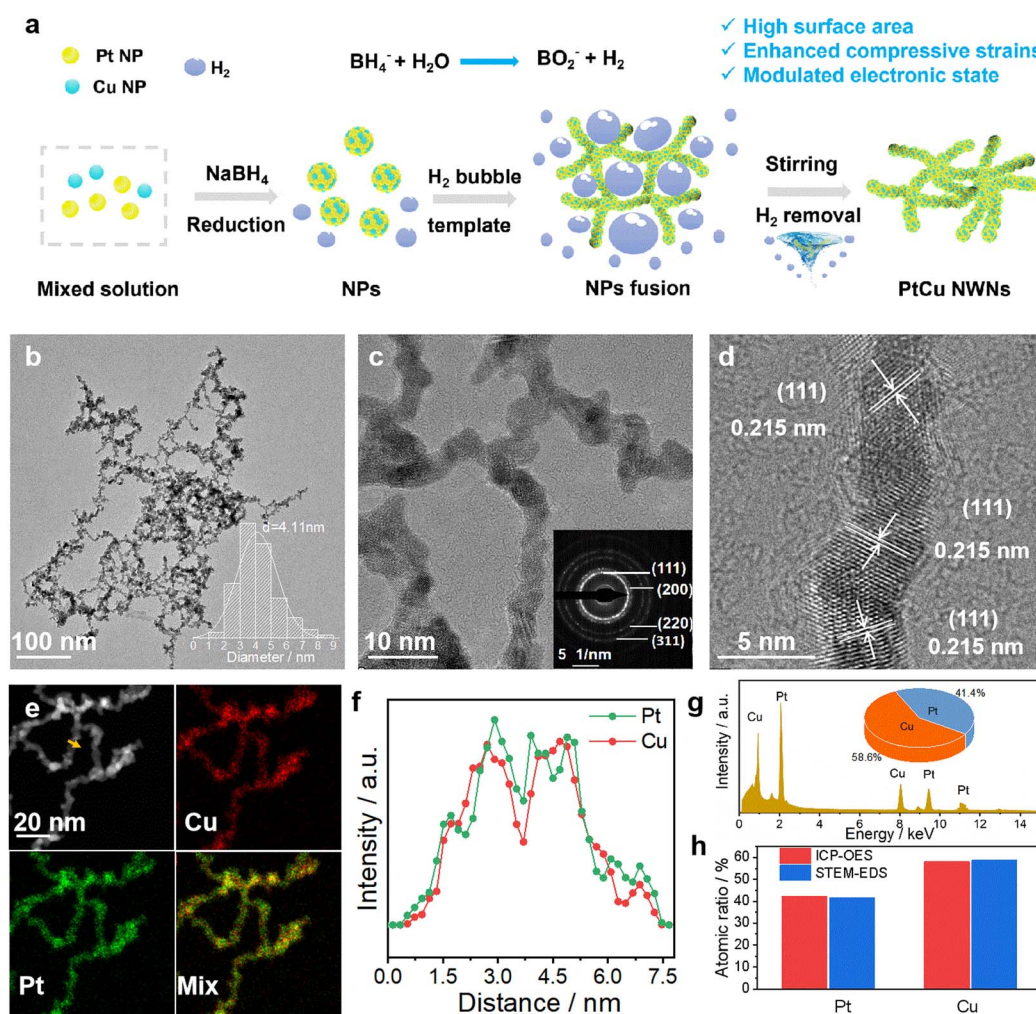


Fig. 1 (a) Schematic illustration of the preparation of Pt_xCu_y NWNs. Structural and compositional characterization of Pt₄₂Cu₅₈ NWNs. Representative (b) low- and (c) high-magnification TEM images (inset shows the corresponding SAED pattern). (d) HRTEM image. (e) HAADF-STEM image and STEM-EDS elemental mappings. (f) Line-scanning profiles across the yellow arrow shown in (e). (g) STEM-EDS spectra. (h) The atomic ratio of Pt₄₂Cu₅₈ NWNs characterized by STEM-EDS and ICP-OES.



reducing agent, and the *in situ* generated hydrogen as the dynamic template for the formation of 1D structures³¹ (details in the Experimental section in the ESI†). In this synthesis, the addition of sodium borohydride at low temperatures is crucial in controlling the kinetics of nucleation and growth of NWNs. Replacement with another strong reducing agent, such as $\text{N}_2\text{H}_4 \cdot \text{H}_2\text{O}$, would result in the formation of nano-dendrites rather than NWs (Fig. S1†), due to the lack of the dynamic template of hydrogen. And the increase in temperature would lead to the formation of low-quality NWNs, due to the accelerated nanoparticle assembly (Fig. S2†).

The morphology of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs obtained from the scanning electron microscopy (SEM) in Fig. S3† and the low-magnification transmission electron microscope (TEM) in Fig. 1b reveal a 3D porous nano-network structure, which is primarily composed of interconnected wavy 1D NWs with a diameter of approximately 4.1 nm. The selected-area electron diffraction (SAED) pattern (inset of Fig. 1c) officially defines the polycrystalline nature of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs. The detailed structural characterization of these NWs was further conducted using high-resolution transmission electron microscopy (HRTEM). HRTEM images reveal that the lattice spacing of the $\text{Pt}_{42}\text{Cu}_{58}$ NWNs is 0.215 nm, which is smaller than that of the surface cubic platinum (fcc) at 0.226 nm. This observation indicates the formation

of an alloy structure with lattice contraction, as shown in Fig. 1d. The (111) crystal planes on the surface of NWNs are not oriented in parallel, which may lead to the development of wavy structures. The homogeneous distribution of Pt and Cu elements throughout the whole $\text{Pt}_{42}\text{Cu}_{58}$ NWN surface was observed in the STEM-EDX elemental mapping images (Fig. 1e). The EDS line-scanning analysis, conducted along the direction indicated by the yellow arrows in Fig. 1e (as shown in Fig. 1f), demonstrated a uniform distribution of Pt and copper Cu. In addition, the atomic ratio of Pt to Cu, as determined by STEM-EDS, was 41.4 : 58.6, which is in agreement with the results obtained from inductively coupled plasma-optical emission spectrometry (ICP-OES) measurements (42.0 : 58.0). Overall, the aforementioned results provide compelling evidence for the successful synthesis of ultrathin cross-linked $\text{Pt}_{42}\text{Cu}_{58}$ NWNs. The catalysts featuring analogous structures but varying Pt/Cu ratios as control samples were also successfully synthesized by adjusting the molar ratio of the two metal precursors. The morphology of the NWNs and the corresponding atomic ratios are presented in Fig. S4–S6 and Table S1.†

2.2 Strain and ligand effects

To investigate the phase and electronic structure of the prepared Pt_xCu_y NWNs, X-ray diffraction (XRD), X-ray

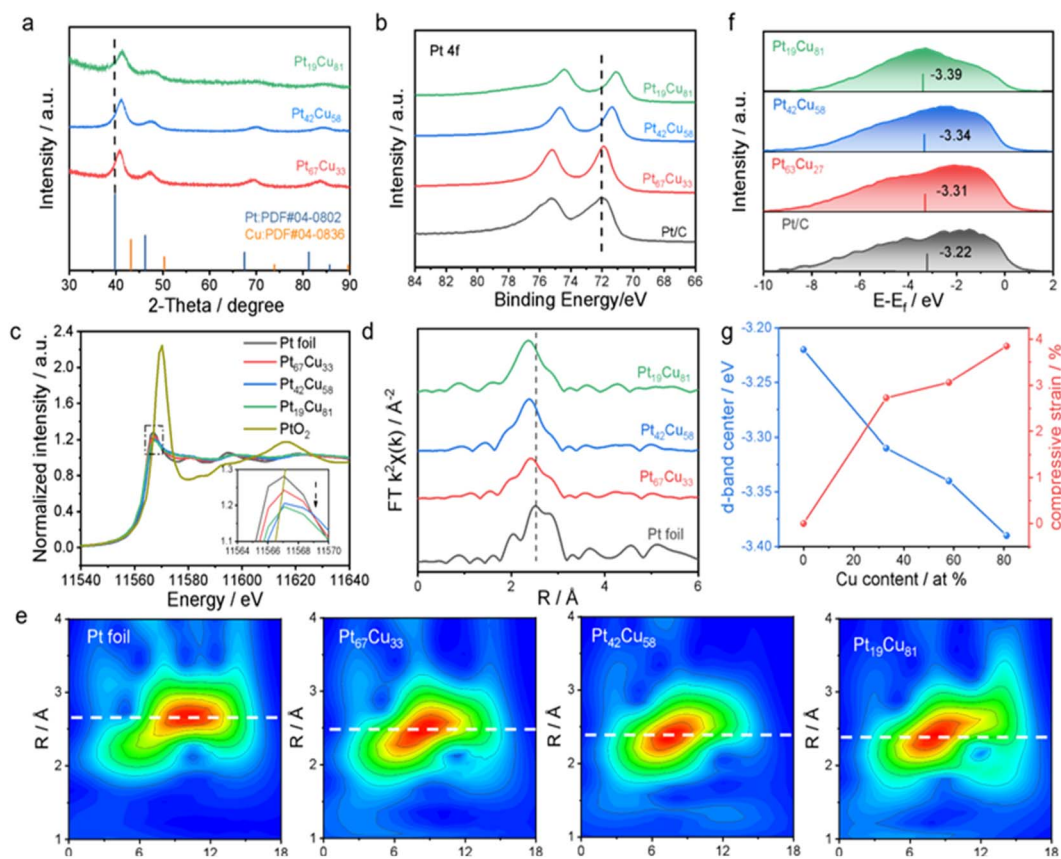


Fig. 2 Characterization of Pt_xCu_y NWNs. (a) XRD pattern of Pt_xCu_y NWNs. (b) High-resolution XPS spectra of Pt 4f for Pt_xCu_y NWNs and commercial Pt/C. (c) Normalized Pt $\text{L}_{3\text{-edge}}$ XANES spectra of Pt_xCu_y NWNs, PtO_2 , and Pt foil, respectively. (d) FT-EXAFS spectra of Pt_xCu_y NWNs and Pt foil. (e) Wavelet transform (WT) plots of Pt_xCu_y NWNs and Pt foil, respectively. (f) Surface valence band photoemission spectra of Pt_xCu_y NWNs and commercial Pt/C. (g) Evolution of the d-band center and lattice strain as a function of Cu content in Pt_xCu_y NWNs.

photoelectron spectroscopy (XPS), and X-ray adsorption fine structure (XAFS) were carried out. The XRD diffraction peaks of Pt_xCu_y NWNs with different compositions (Fig. 2a) show a typical fcc polycrystalline structure and are located between the standard diffraction spectra of Pt metal (JCPDS card no. 04-0802) and Cu (JCPDS card no. 04-0836), indicating the formation of a PtCu binary alloy. The diffraction peak positions shift to higher angles with increasing Cu content relative to pure Pt, further confirming the presence of compressive strain due to lattice mismatch between Pt and the smaller atomic radius Cu. As shown in Table S2,[†] the compressive strains induced in the Pt (111) lattice of $\text{Pt}_{67}\text{Cu}_{33}$, $\text{Pt}_{42}\text{Cu}_{58}$, and $\text{Pt}_{19}\text{Cu}_{81}$ are 2.73%, 3.06%, and 3.85%, respectively.^{32–35}

XPS analysis revealed the electron transfer from Cu to Pt due to the difference in electronegativity (Fig. 2b and S10[†]). As shown in Fig. 2b, the binding energy of the Pt 4f peak shifts negatively with lower electronegativity Cu doping and that of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs demonstrates a 0.69 eV negative shift compared to commercial Pt/C (Fig. S10[†]), which can lower the platinum d-band center for the increased number of filled orbitals.^{18,36} Concurrently, the Cu 2p spectra (Fig. S11[†]) reveal a gradual positive shift in binding energy with higher Cu content, confirming a reduction in electron density of Cu atoms. This complementary trend in binding energy for Pt 4f and Cu 2p directly validates the electron donation from Cu to Pt. Besides, we can also find that most of the Pt and Cu on the surface of $\text{Pt}_{42}\text{Cu}_{58}$ were in the metallic state (Fig. S10[†]), demonstrating the formation of the alloy.

Fig. 2c presents the normalized Pt L_3 -edge X-ray absorption near-edge structure (XANES) spectra of Pt_xCu_y NWNs compared with Pt foil and PtO_2 . They exhibit a high degree of similarity to that of Pt foil, with a notable divergence from that of PtO_2 , demonstrating the Pt is mainly in a metallic state.^{37,38} As previously reported, the intensity of the white line is proportional to the transition of electrons from the Pt $2p_{3/2}$ core to the 5d state, and it can be used as an indicator of the electron occupancy of the Pt 5d band.²⁰ A lower white line peak intensity is observed in Pt_xCu_y NWNs compared to that in Pt foil. By integrating the Pt L_3 -edge white line areas,^{39,40} we quantified the 5d band holes (vacancies) as 0.785, 0.771, and 0.745 for $\text{Pt}_{67}\text{Cu}_{33}$, $\text{Pt}_{42}\text{Cu}_{58}$, and $\text{Pt}_{19}\text{Cu}_{81}$, respectively, systematically lower than those of pure Pt (1.000) and demonstrating a Cu-dependent reduction trend (Fig. S7, S8 and Table S3[†]). This progressive decline in 5d vacancies signifies enhanced electron density within the Pt 5d orbitals upon Cu incorporation, corroborating our XPS findings. This electron enrichment lowers the Pt d-band center, thereby weakening adsorbate binding strength.³⁶ The Fourier transform extended XAFS (EXAFS) (Fig. 2d) and its fitting results (Fig. S9 and Table S3[†]) clearly confirmed the coordination environment of the Pt L_3 edge for the catalysts. As shown in Fig. 2d, the k^2 -weighted EXAFS for Pt_xCu_y NWNs at Pt L_3 -edge without phase correction shows a lower scattering peak ($\text{Pt}_{67}\text{Cu}_{33}$ at 2.479 Å, $\text{Pt}_{42}\text{Cu}_{58}$ at 2.446 Å, and $\text{Pt}_{19}\text{Cu}_{81}$ at 2.402 Å) in contrast to Pt foil (2.632 Å), which could be assigned to the Pt–Pt/Cu paths, confirming that the Pt_xCu_y NWNs predominantly exhibit shortened Pt–M (M = Pt/Cu) metal bonds. This observation indicates the formation of

the PtCu alloy structure with lattice contraction. Meanwhile, the fitting results in Table S3[†] can also prove the existence of Pt–Pt and Pt–Cu bonds in the PtCu NWNs. For instance, the main metal bonds are the Pt–Cu bond with a coordination number of 4.1 and the Pt–Pt bond with a coordination number of 5.0 in $\text{Pt}_{42}\text{Cu}_{58}$ NWNs (Table S4[†]). There are no Pt–O bonds in the Pt_xCu_y NWNs compared to PtO_2 , which is also visually confirmed by the wavelet transform (WT) (Fig. 2e and S12[†]). Furthermore, it is noteworthy that the bond length of the Pt–M bond (about 2.65–2.72 Å) in *R* space has been shortened with more Cu content involved in comparison to the Pt–Pt bond (2.77 Å) in Pt foil from the EXAFS and WT analysis, demonstrating the compressive strain involved, which is in agreement with the HRTEM and the XRD results.⁴¹ The compressive strain can broaden the d-band and then downshift the d-band center, resulting from the increasing orbital overlap in Pt atoms after smaller metal atoms are involved.^{42,43}

Prior research has demonstrated that a negative shift in the d-band center can weaken the binding energy between the catalyst surface and adsorbates.^{44,45} The findings illustrated above indicate that Pt_xCu_y NWNs, which exhibit a rich-electron Pt surface and existing compressive strain, can synergistically decrease the d-band center, causing it to deviate from the Fermi level. Therefore, the XPS valence band spectrum of the catalysts was measured and analyzed (Fig. 2f). The results show that the d-band center positions of the compressive strained $\text{Pt}_{67}\text{Cu}_{33}$ NWNs, $\text{Pt}_{42}\text{Cu}_{58}$ NWNs, and $\text{Pt}_{19}\text{Cu}_{81}$ NWNs have decreased by 0.09 eV, 0.12 eV, and 0.17 eV, respectively, compared to that of Pt/C, suggesting that Cu incorporation leads to a downward shift in the d-band center, which may enhance the MOR activity by weakening the binding force between the Pt-based catalysts and the carbonyl intermediate species adsorbed on the surface. Furthermore, a significant positive correlation was observed among the compressive strain, the d-band center, and the Cu content (Fig. 2g). This indicates that an increase in Cu content corresponds to a greater compressive strain and a lower d-band center. Thus, the compressive strain and d-band electrons can be effectively tuned by merely modifying the Cu content, allowing for the regulation of intermediate adsorption strength and the optimization of catalyst performance for the MOR.

2.3 Electrocatalytic performances of the Pt_xCu_y NWNs

The MOR activity and stability of the obtained Pt_xCu_y NWNs and commercial Pt/C were evaluated under acid conditions. The details of the tests were in the Experimental section in the ESI.[†] The electrochemically active surface area (ECSA) was measured by CV in acid electrolyte at 50 mV s^{−1} (Fig. 3a). The calculated ECSA values for $\text{Pt}_{67}\text{Cu}_{33}$, $\text{Pt}_{42}\text{Cu}_{58}$, $\text{Pt}_{19}\text{Cu}_{81}$, and commercial Pt/C are 29.12, 30.09, 33.10, and 37.75 m² g_{Pt}^{−1}, respectively. Besides, the ECSA values of the three Pt_xCu_y NWNs are comparable to those of Pt/C, indicating that the 3D nano-networks with ultrathin NWs could effectively expose active sites and improve atomic utilization.

The MOR curves of catalysts were recorded in an acid electrolyte containing methanol (see Fig. 3b and c). As illustrated in Fig. 3d, the $\text{Pt}_{42}\text{Cu}_{58}$ NWNs demonstrate the highest mass



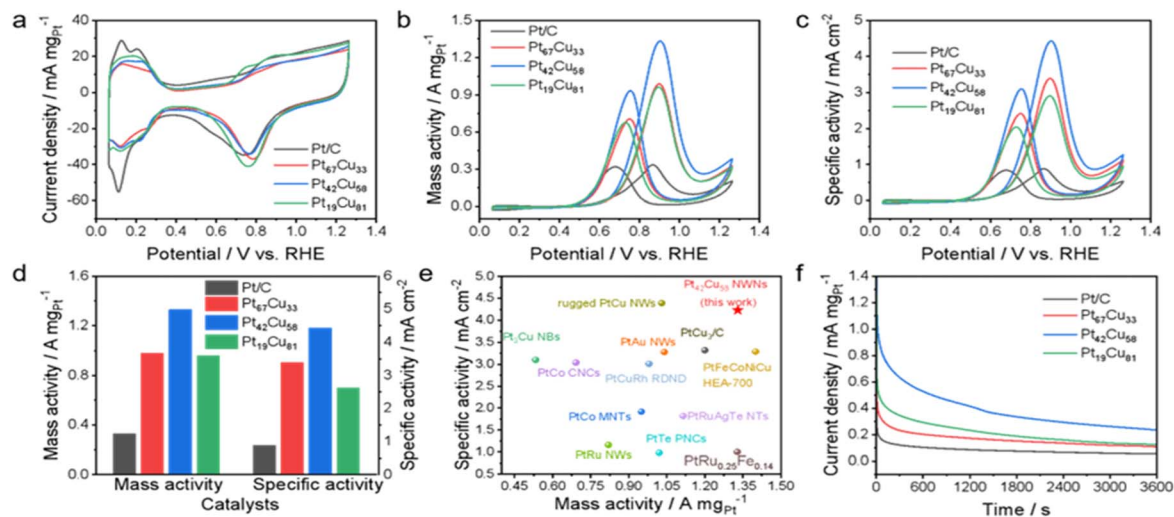


Fig. 3 Electrochemical performances of Pt_xCu_y NWNs and commercial Pt/C for the MOR. (a) CV curves of the catalysts recorded in N_2 -saturated 0.1 M HClO_4 electrolyte at a scan rate of 50 mV s^{-1} . (b) Pt mass-normalized and (c) ECSA-normalized MOR curves measured in 0.1 M HClO_4 containing 0.5 M methanol solution at a scan rate of 50 mV s^{-1} . (d) Histogram of MOR mass activities and specific activities of the corresponding catalysts. (e) Comparison of MOR activities of other reported Pt-based electrocatalysts. (f) $i-t$ curves from the chronoamperometric tests in 0.1 M HClO_4 containing 0.5 M methanol solution.

activity (MA, $1.33 \text{ A mg}_{\text{Pt}}^{-1}$), which is 1.36, 1.39, and 4.03 times higher than those of $\text{Pt}_{67}\text{Cu}_{33}$ NWNs ($0.98 \text{ A mg}_{\text{Pt}}^{-1}$), $\text{Pt}_{19}\text{Cu}_{81}$ NWNs ($0.96 \text{ A mg}_{\text{Pt}}^{-1}$), and commercial Pt/C ($0.33 \text{ A mg}_{\text{Pt}}^{-1}$), respectively. Furthermore, the specific activity (SA, 4.43 mA cm^{-2}) of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs obtained by normalizing the ECSA is also higher than those of $\text{Pt}_{67}\text{Cu}_{33}$ NWNs (3.39 mA cm^{-2}), $\text{Pt}_{19}\text{Cu}_{81}$ NWNs (2.63 mA cm^{-2}), and commercial Pt/C (0.88 mA cm^{-2}), respectively. A significant volcano-type dependence on MOR activity was observed, influenced by the Cu component, compressive strain, and the position of the d-band center (Fig. S13†). The MOR activity of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs is compared with that of previously reported Pt-based catalysts, as illustrated in Fig. 3d and summarized in Table S5.† The enhanced MOR activity may be attributed to the optimized electron structure of Pt, which has been tuned by strain and ligand effects after Cu doping and the presence of enriched active sites for the ultra-thin 3D structure.

The catalytic stability was evaluated through chronoamperometry (CA) tests. As illustrated in Fig. 3d, the current densities for $\text{Pt}_{42}\text{Cu}_{58}$ NWNs remained superior to those of the other catalysts throughout the 3600 s CA tests. The structural change of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs was negligible, except for a slight widening of the lattice stripes from 0.215 nm to 0.218 nm, which may result from the unavoidable leaching of Cu atoms during the reaction process (Fig. S14†). In contrast, the activity of Pt/C was poor, accompanied by severely damaged morphology due to CO poisoning and severe aggregation of the nanoparticles (Fig. S15†). The electrochemical tests above demonstrate that the excellent MOR catalytic activity and stability of $\text{Pt}_{42}\text{Cu}_{58}$ NWNs can be attributed to the robust CO tolerance after Cu doping and its advanced 3D structure.

To achieve a deeper understanding of the catalysts' activity, a series of kinetic studies were undertaken. Linear sweep

voltammetry (LSV) measurements were conducted in acidic methanol media. As shown in Fig. S16a and b,† $\text{Pt}_{42}\text{Cu}_{58}$ NWNs have the lowest onset potential and Tafel slope ($80.45 \text{ mV dec}^{-1}$) compared to the other catalysts, indicating their enhanced MOR kinetics. As shown in Fig. S16c and Table S6,† $\text{Pt}_{42}\text{Cu}_{58}$ NWNs also possess the highest values of j_0 and α , both of which are calculated from the Tafel equation, demonstrating accelerated electron transfer.⁴⁶ The electrochemical impedance spectroscopy (EIS) analysis was further performed, as shown in Fig. S16d.† The impedance arc diameter of the $\text{Pt}_{42}\text{Cu}_{58}$ NWNs is significantly smaller compared to that of the other catalysts, indicating a lower charge transfer resistance and enhanced kinetic activity. These findings are in agreement with the previously mentioned results. Additionally, LSV curves at various scan rates were obtained for all samples (Fig. S17†). The plots of current density *versus* the square root of the scan rate for all electrocatalysts demonstrate a linear relationship, suggesting a diffusion-limited process.⁴⁷ The $\text{Pt}_{42}\text{Cu}_{58}$ NWNs show the highest slope of 0.52, demonstrating accelerated mass/electron transfer.⁴⁶ Finally, we also obtained LSV profiles at different temperatures to investigate the thermodynamics of the MOR for all samples. Fig. S18† illustrates the linear relationship between the current density $\ln j$ and $1/T$ for all the samples. The corresponding apparent activation energy values (E_a) calculated from the slope value of the Arrhenius plot can reflect the adsorption energy of methanol and intermediate species on the catalyst surface.⁴⁸ The $\text{Pt}_{42}\text{Cu}_{58}$ NWNs show the smallest value of E_a ($39.93 \text{ kJ mol}^{-1}$) compared to the other catalysts, indicating that methanol oxidation is more facile for $\text{Pt}_{42}\text{Cu}_{58}$ NWNs.⁴⁹ In brief, $\text{Pt}_{42}\text{Cu}_{58}$ NWNs that achieve an optimal balance between the strain and ligand effects can substantially reduce the energy barrier and enhance the kinetics of the MOR.



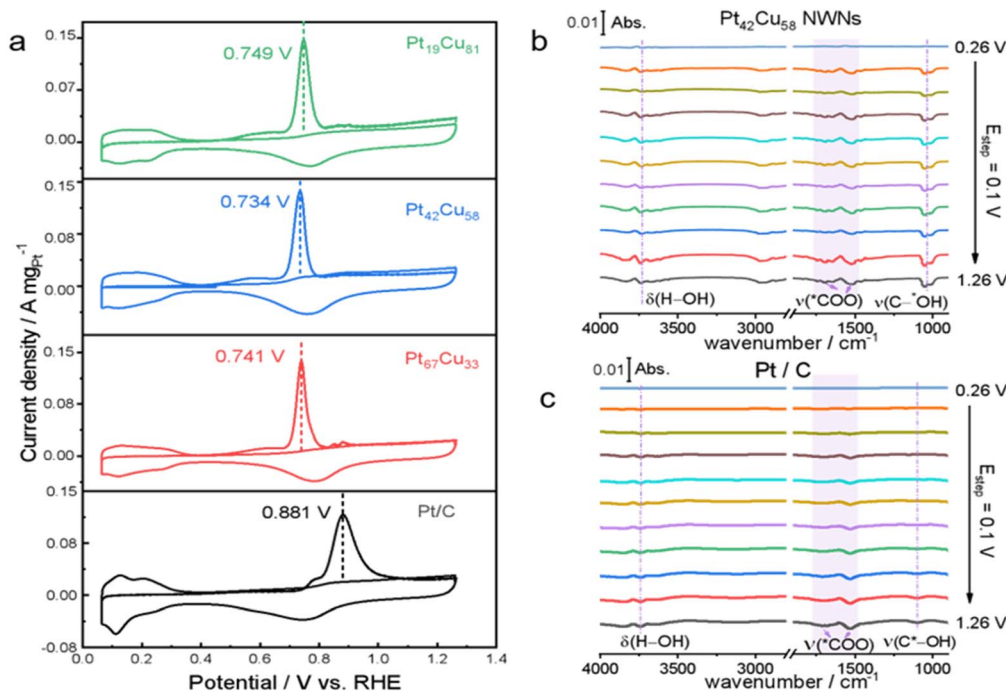


Fig. 4 (a) CO stripping tests of Pt_xCu_y NWNs and commercial Pt/C. (b) Electrochemical *in situ* ATR-FTIR spectrum of the MOR on Pt₄₂Cu₅₈ NWNs and (c) Pt/C.

2.4 Anti-CO poisoning ability of the Pt_xCu_y NWNs

To further explore the anti-CO poisoning ability of Pt₄₂Cu₅₈ NWNs in the enhancement of MOR performance, we employed CO stripping and *in situ* attenuated total reflectance Fourier transform infrared (ATR-FTIR) spectroscopy. CO is a key intermediate in the MOR process.⁵⁰ As shown in Fig. 4a, the CO stripping peak potentials for all Pt_xCu_y alloys shift negatively in comparison to commercial Pt/C (0.881 V vs. RHE), indicating improved CO oxidation capabilities and increased resistance to CO poisoning. Notably, the Pt₄₂Cu₅₈ NWNs (0.734 V vs. RHE) display a lower peak potential than the Pt₁₉Cu₈₁ NWNs (0.749 V vs. RHE) and Pt₆₇Cu₃₃ NWNs (0.741 V vs. RHE), suggesting that a moderate downshift in the d-band center is crucial for facilitating the removal of CO and enhancing the capacities for CO tolerance.

ATR-FTIR spectroscopy was further used to detect the adsorbed species at the molecular level on the optimized Pt₄₂Cu₅₈ NWNs and commercial Pt/C for comparison (Fig. 4b and c). We can find featured bonds like the stretching vibration C*-OH (about 1022 cm⁻¹) and the antisymmetric stretching vibration *COO (from 1500 to 1630 cm⁻¹), which belong to methanol and the intermediate *COOH, respectively.^{5,51,52} With the potential improved, those bonds appeared first at 0.36 V in Pt₄₂Cu₅₈ NWNs, 100 mV lower than that of commercial Pt/C (0.46 V), implying the high methanol activation capacity of Pt₄₂Cu₅₈ NWNs. The stretching bonds O-H (about 3700 cm⁻¹) detected in the two samples may be assigned to OH* from the water dissociation,⁵³ which can oxidize the CO* to form *COOH and then dehydrogenate to CO₂.¹⁴ The combined results of CO stripping and *in situ* ATR demonstrate that the Pt₄₂Cu₅₈ NWNs exhibit a high anti-CO poisoning ability and methanol

activation capacity. This may be attributed to the weak adsorption of CO on Pt sites, which facilitates the removal of CO and then the liberation of active sites, thereby promoting methanol activation.

3 Conclusions

In conclusion, we designed a kind of compressive strained Pt₄₂Cu₅₈ NWN with enhanced catalytic activity and robust anti-CO poisoning capacity for the MOR. The experiments demonstrated that the synergy of compressive strain and the ligand effect endowed the catalysts with a downward shifted d-band center, which can weaken the CO adsorption strength on the Pt surface. The optimized Pt₄₂Cu₅₈ NWNs achieve a high MOR mass activity of 1.33 A mg_{Pt}⁻¹ and specific activity of 4.43 mA cm⁻², which are 4.03 and 5.03 times higher than those of Pt/C, respectively. The CO stripping and *in situ* ATR-FTIR results further indicate their enhanced CO tolerance ability and methanol activation capacity. Moreover, the Pt₄₂Cu₅₈ NWNs exhibit excellent stability with high current densities observed after 3600 s of continuous operation attributed to their superior resistance to CO poisoning and the stable 3D NWN structure. This work provides a feasible strategy to suppress CO poisoning during methanol oxidation electrocatalysis and obtain highly efficient anode catalysts with enhanced durability in the DMFC field.

Data availability

All data generated or analyzed during this study are included in this article and the ESI.†



Author contributions

All of the authors contributed to the manuscript preparation. Shiyue Xing, Huihui Li and Chunzhong Li conceived the outline of the manuscript. Shiyue Xing, Huihui Li and Chunzhong Li wrote the original draft of the manuscript. Zhongliang Liu, Yingfang Jiang, Pinghui Tang, Jian Zhang, and Jiatang Chen discussed and helped revise the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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