



Cite this: *Chem. Sci.*, 2025, **16**, 8069

All publication charges for this article have been paid for by the Royal Society of Chemistry

Received 23rd January 2025
Accepted 28th March 2025

DOI: 10.1039/d5sc00610d
rsc.li/chemical-science

Introduction

Noble metal nanoparticles (NPs) are well known for their capacity to enhance the interactions between light and matter, exhibiting fascinating colors.¹ For an individual particle, the collective oscillation of its free electrons results in a localized surface plasmon resonance (LSPR) at a certain frequency determined by the size/shape of the NP and its dielectric environment. When two or more NPs are close enough to each other, the LSPR of the individual NPs interacts through their optical near fields, forming coupled oscillation modes and generating a significant change in the optical response.^{2,3} Furthermore, coupling systems with more complex structures exhibit some strikingly different optical properties, such as plasmon hybridization,⁴ Fano resonance,⁵⁻⁷ electromagnetically induced transparency (EIT),⁸ and plasmonic waveguiding,⁹ which are widely applied from sensing¹⁰ to energy harvesting,^{11,12} and subwavelength optical imaging.¹³ Thus, plasmon

coupling is the fundamental principle by which the optical resonances in NP assemblies are tuned. Moreover, plasmon coupling of the hybrid nanostructure can induce significant electromagnetic field enhancement, which is beneficial for plasmon-enhanced spectroscopy,^{14,15} artificial photosynthesis,¹⁶ and plasmon-enhanced catalysis.¹⁷⁻¹⁹

In the face of the increasingly serious energy crisis, one expects to maximize the conversion of solar energy into usable chemical energy through plasmon-enhanced photocatalysis.²⁰ Noble metal NPs combined with semiconductors were developed to form heterostructures where plasmonic metal acts as antennae, enhancing the light absorption and photogenerated carrier lifetime in semiconductors greatly.²¹⁻²⁴ Recent advances in tuning the surface/interfacial structures of the hybrid plasmonic nanocatalysts have shown great promise for their application in photocatalysis.²⁵⁻²⁹ Notably, strong coupling can enhance the interaction between light and matter, which has been employed to modulate the catalytic activities and selectivity. For instance, the coupling between the LSPR and optical modes such as whispery gallery modes,³⁰ waveguide modes,^{9,31} and Bloch waves³² has been demonstrated for broadband optical responses. In particular, Misawa and coworkers demonstrated that coupling between the LSPR mode and Fabry-Pérot (F-P) nanocavity mode has been shown to achieve optical responses over a broad range of wavelengths, which can promote water splitting.³³ In addition, the coupling strength

^aCollege of Chemistry and Molecular Sciences, Hubei Key Laboratory of Electrochemical Power Sources, Wuhan University, Wuhan 430072, China. E-mail: lichaosun@whu.edu.cn; zhangqf@whu.edu.cn

^bSuzhou Institute of Wuhan University, Suzhou 215123, China. E-mail: lichaosun@whu.edu.cn

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d5sc00610d>



can be tuned by varying the dielectric constant and the embedding depth of NPs.^{34,35} However, limitations still exist in the understanding of the coupling between optical modes in individual 3D structures. The core–shell–satellite structures have been constructed in some previous studies for plasmon-enhanced nanocatalysis;^{36,37} however, the quantitative correlations between structural features and modal coupling in hybrid structures remain a challenge yet to be addressed. Moreover, how the modal coupling in hybrid structures correlated with their activities in enhancing plasmonic photocatalysis remains unclear. Therefore, tuning the modal coupling in 3D hybrid plasmonic structures through precise control over geometric features and quantitative optical characterization studies is urgent for developing highly efficient plasmonic photocatalysts.

Herein, we present an $\text{Au}@\text{Cu}_2\text{O}@\text{Au}$ (ACA) core–shell–satellite hybrid structure with tunable plasmon coupling for enhanced plasmon photocatalysis. In this structure, $\text{Au}@\text{Cu}_2\text{O}$ NPs support a nanocavity mode in the visible spectral region. Furthermore, the growth of Au NPs as satellites with tailored sizes induces a modal splitting in the extinction spectra, suggesting the formation of a modal coupling. By deliberately controlling the thickness of the Cu_2O shell and the size of Au satellites, the coupling can be tuned and understood using a two-harmonic resonator model. The modal splitting in scattering spectra was also revealed through single-particle dark-field scattering (DFS) measurements. More importantly, the ACA NPs show much better catalytic activities than $\text{Cu}_2\text{O}@\text{Au}$ NPs toward plasmon-driven chemical reactions revealed by *in situ* surface-enhanced Raman scattering (SERS) measurements. Ultrafast transient absorption (TA) spectroscopy and photoelectrochemical (PEC) measurements further show the superior performance of the ACA hybrid structure toward plasmonic photocatalysis.

Results and discussion

As schematically illustrated in Fig. 1A, we developed a two-step method for the construction of the ACA core–shell–satellite nanostructure (see experimental details in the ESI†). Au rhombicuboctahedral (RCO) NPs, which are single-crystalline in nature and ~55 nm in size, were employed as the plasmonic core (Fig. 1B and S1†). Firstly, the Cu_2O shell was grown on the surfaces of the Au NPs to form the $\text{Au}@\text{Cu}_2\text{O}$ structure with tunable shell thickness (Fig. 1C and S1, S2†). The range of Cu_2O shell thickness in $\text{Au}@\text{Cu}_2\text{O}$ NPs can be tuned from ~27 nm to ~65 nm by simply changing the amount of Cu^{2+} precursors (Fig. S2†). Secondly, ACA NPs can be obtained by reacting the Cu_2O shell with HAuCl_4 through a galvanic replacement reaction (GRR) process (Fig. 1D and S3†). The core–shell–satellite structure can be then clearly visualized by HAADF-STEM imaging and corresponding EDS mapping of an individual ACA particle (Fig. 1E). X-ray diffraction (XRD) patterns further verify the composition of Au and Cu_2O in the ACA NPs and none of the Cu^0 elements or CuAu alloy phase can be found (Fig. S4†). By deliberately controlling the amount of HAuCl_4 , the size and density of the Au satellite in ACA NPs can be finely modulated, resulting in a different modal coupling mode that will be

discussed later (Fig. S5†). The as-prepared ACA NPs exhibit excellent uniformity in both the overall size of ACA NPs and the density of the Au satellites, which will benefit the investigations of plasmon coupling in the ensemble extinction spectra (Fig. S6†). In the ACA structure, Cu_2O not only acts as a medium allowing light to spread but also as a spacer to control the distance between the cavity and Au satellites, which is similar to the spacer of NPs on thin film systems.³³ Moreover, the *in situ* GRR process could create close contacts between Au satellites and the Cu_2O shell (Fig. S5†), which could potentially enhance the plasmon coupling in contrast to the method that used long alkyl chain ligands as the linker to assemble the Au satellites.³⁸ Notably, the Au core could serve as an Au mirror in the ACA structure, which could improve the optical distance effectively when light undergoes repeated mirror reflection in the nanostructure. Therefore, it is envisioned that the ACA hybrid structure provides the superb capability to confine the electromagnetic field of light into a dielectric layer and enhance the coupling between light and matter vastly.

The formation of a core–shell–satellite coupled structure introduced intriguing modifications to the optical properties. Fig. 1F shows the extinction spectra of Au, $\text{Au}@\text{Cu}_2\text{O}$, and ACA NPs in aqueous solution. Au NPs exhibit a dipole resonance peak localized at ~530 nm, whereas $\text{Au}@\text{Cu}_2\text{O}$ NPs show a dipole resonance peak located at ~685 nm. Interestingly, the original dipole resonance peak of $\text{Au}@\text{Cu}_2\text{O}$ splits into two peaks after the formation of Au satellites on the surface of $\text{Au}@\text{Cu}_2\text{O}$ NPs. The Gaussian or Lorentzian fittings of the dual extinction bands of ACA NPs further show the peak splitting in the extinction spectrum of ACA NPs (Fig. 1G and S7†). The newly fitted peaks are labeled as ω^+ and ω^- , which are localized at ~590 nm and ~690 nm, respectively. Inspired by the modal strong coupling system developed by Misawa and coworkers,³³ we proposed that the observed peak splitting in our case is akin to the energy-level splitting into upper and lower modes generated by the modal coupling (Fig. 1H). Note that Fig. 1H shows the enhanced light reflection and absorption in the ACA structure but not a whispering gallery mode (WGM). Energy-level diagram of the coupling between the nanocavity mode in the Cu_2O layer and the LSPR mode of the Au satellite NPs was showed in the right panel of Fig. 1H. The ω_{sp} and ω_{cavity} are the resonant frequencies of plasmon and nanocavity modes. The LSPR mode of Au satellites is thought to be coupled with the nanocavity mode of the $\text{Au}@\text{Cu}_2\text{O}$ layer in the ACA structure upon light excitation, leading to the peak splitting in the ensemble extinction spectrum. Although the nanocavity mode here is not a traditional F-P cavity mode as Misawa and coworkers demonstrated in their cases,^{33,34} the inner $\text{Au}@\text{Cu}_2\text{O}$ core in the ACA structure could be considered as a nanocavity mode when interacting with light and the Au satellites for enhanced light absorption. To satisfy the strong coupling regime between two oscillators, the frequency of the coupled system should be higher than the dephasing time of both oscillators.^{39,40} However, it's very difficult for us to estimate the splitting energy on account of the thickness change of Cu_2O and the variation in Au satellite size during the process of loading Au satellites, so we couldn't obtain meaningful results such as the



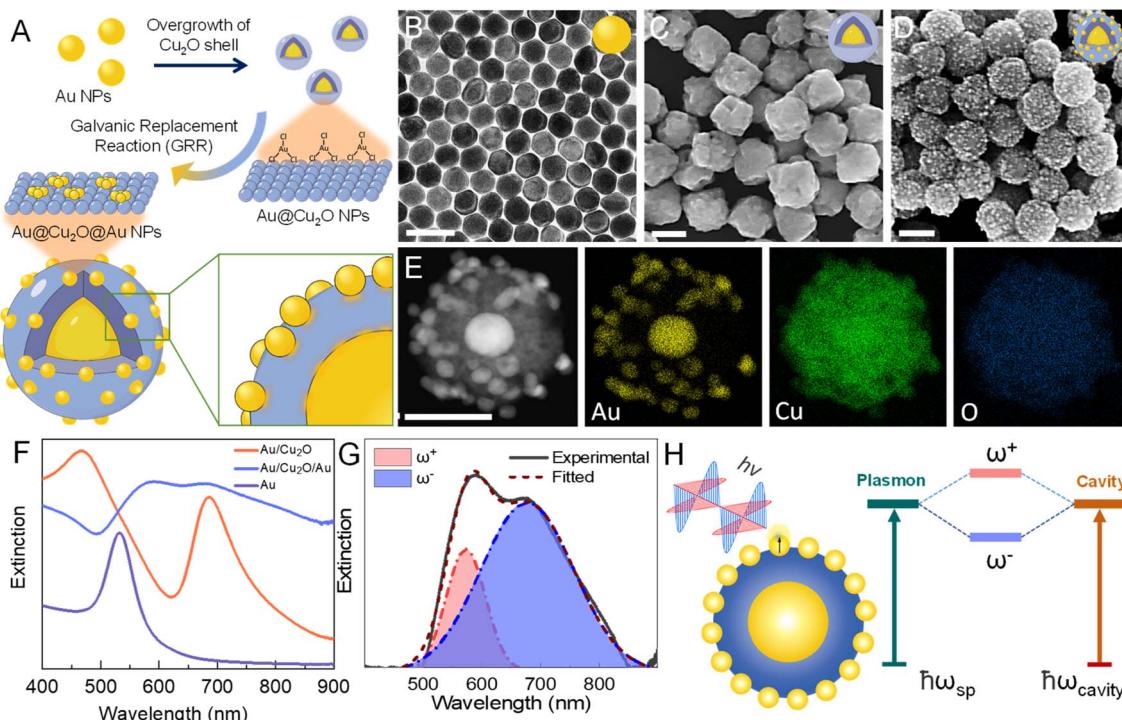


Fig. 1 Construction and characterization of $\text{Au}@\text{Cu}_2\text{O}@\text{Au}$ (ACA) core–shell–satellite nanostructures with modal splitting in the extinction spectra. (A) Schematic illustration of the construction of the ACA nanostructure. (B) TEM image of Au NPs. (C) SEM image of $\text{Au}@\text{Cu}_2\text{O}$ NPs. (D) SEM image of the ACA NPs with smaller Au satellites. (E) STEM image of ACA NPs and the corresponding EDS mapping images. The yellow, green, and blue colors depict Au, Cu, and O elements, respectively. (F) Extinction spectra of Au, $\text{Au}@\text{Cu}_2\text{O}$, and ACA NPs in aqueous solution. (G) Extinction spectra of ACA NPs fitted using a Gaussian function. The red and blue curves indicate the Gaussian fitting of the dual extinction bands. (H) Schematic illustration of the interaction between incident light and the ACA structure; an energy-level diagram of the modal coupling between the cavity mode in the Cu_2O layer and the LSPR mode of the Au satellites NPs. The $\hbar\omega_{\text{sp}}$ and $\hbar\omega_{\text{cavity}}$ are the resonant frequencies of plasmon and nanocavity modes. Scale bars: 100 nm.

inverse cross curve of strong coupling.³³ Therefore, we use a modal coupling mechanism to account for the optical properties that we observed on the ACA structure and further propose another system for understanding the plasmon coupling in the three-dimensional ACA hybrid structures.

For the plasmonic NPs with a coupling system, it is well known that the size, permittivity, shape, and distance between NPs determine the optical properties of the coupled system.^{41–45} Thus, we further tuned the structural features and plasmon coupling of ACA NPs by controlling the size and density of Au satellites (Fig. 2). By simply changing the amount of Au precursors involved in the GRR with the Cu_2O shell, ACA NPs with tunable size and density of Au satellites can be obtained (Fig. 2C–H and S8†). The thickness of the Cu_2O shell in ACA NPs is found to gradually decrease as the amount of HAuCl_4 increases (Fig. S8†). As shown in Fig. 2A, a gradual broadening of the dipole resonance peak of ACA NPs can be observed as the size and density of Au satellites first increase. A further increase in the size of Au satellites generates a shoulder peak (~ 600 nm), which may indicate the formation of dual-extinction bands due to the peak splitting induced by modal coupling (Fig. S9†). The shoulder peak becomes stronger, so eventually, a broad extinction band was observed (Fig. 2A: F curve), suggesting the superposition of two peaks. As we proposed in Fig. 1H, these

two peaks are classified as ω^+ and ω^- , indicating modal coupling between the nanocavity mode and LSPR mode of Au satellites. As the amount of HAuCl_4 further increases, the ω^+ mode blueshifts and the ω^- mode redshifts, ultimately exhibiting two obvious individual peaks (Fig. 2A: G curve). Also, the extinction band of Cu_2O located at ~ 470 nm decreases gradually during the GRR between the Cu_2O shell and HAuCl_4 , suggesting the changes in the thickness of the Cu_2O shell in ACA NPs.

To gain more quantitative insights into the optical properties of coupled structures, we calculated the extinction spectra of ACA NPs with different densities and sizes of Au satellites using the finite-difference-time-domain (FDTD) method (Fig. 2B). The simulation models were constructed according to the TEM images of a single particle (Fig. 2C–H). The simulated results are in excellent agreement with our experimental observations, indicating that the coupling between the LSPR of Au satellites and the nanocavity mode of the $\text{Au}@\text{Cu}_2\text{O}$ layer strongly affects their optical properties through peak splitting. Notably, the aggregated larger satellites could potentially affect the optical properties of ACA NPs. The plasmon–plasmon coupling between Au satellites cannot be completely excluded for sure. Our simulation models, as shown in Fig. 2, have already accounted for the aggregated states for larger Au satellites.

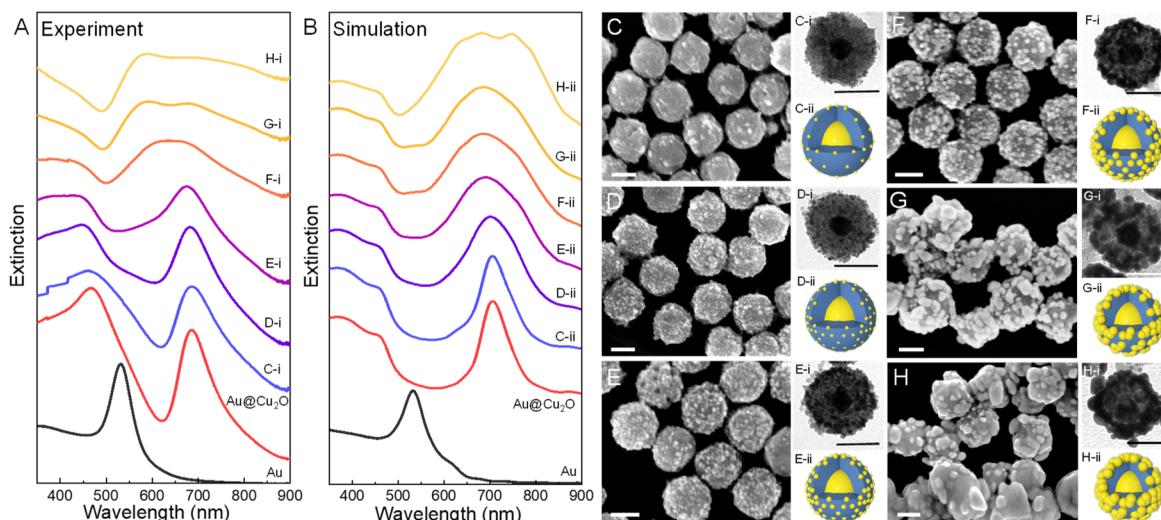


Fig. 2 Tuning the structural features and plasmon coupling of ACA NPs by controlling the size and density of Au satellites. (A) Experimental extinction spectra of Au NPs, Au@Cu₂O NPs (the dipole resonance peak located at 685 nm), and ACA NPs with distinct structures synthesized using different amounts of Au precursors. (B) Calculated extinction spectra of Au NPs, Au@Cu₂O NPs, and ACA NPs. The simulation models were constructed according to the TEM image of a single particle from C-i to H-i. (C–H) SEM and TEM images of ACA NPs with distinct structures, which correspond to the extinction spectra in panel A. Scale bars: 100 nm.

However, it is very difficult to experimentally control the density of aggregated Au satellites for a quantitative comparison. Thus, to quantitatively show the impact of plasmon–plasmon coupling between Au satellites on the optical properties of ACA NPs, we have further carried out the FDTD simulations on ACA NPs with discrete and aggregated Au satellites, respectively. As shown in Fig. S10,† the aggregated Au satellites with a size of 8 nm have a negligible impact on the overall optical properties of ACA NPs. In contrast, for Au satellites with a size of 15 nm, broadening of the plasmon peaks of ACA NPs can be observed as the number of aggregated Au satellites increases. The dependence of optical properties on the coupling between Au satellites and the Au@Cu₂O layer can also be observed and tuned in ACA NPs using anisotropic Au NPs as initial seeds instead of spherical Au NPs (Fig. S11†).

We further investigated the impact of the Cu₂O shell on the modal splitting in the extinction spectra of ACA NPs by varying the thickness of the Cu₂O layer (Fig. 3). We constructed ACA NPs with 5 different Cu₂O thicknesses (the dipole resonance peak located at 630, 650, 667, 685, and 695 nm, respectively). As shown in Fig. 2A and S12,† distinct spectral evolution of extinction bands of ACA NPs with different Cu₂O thicknesses can be observed as a function of the amount of Au precursor. To further qualitatively investigate the spectral differences in different shell thicknesses, two Au@Cu₂O NPs with their dipole resonance peaks located at 630 nm and 667 nm were chosen for a detailed analysis of modal splitting (Fig. 3A and B). The red and blue curves indicate the Gaussian fitting of the dual extinction bands (ω^- and ω^+), which clarifies the transient state from overlapping to the separation of dual-extinction bands. For the peak locations, ω^+ blue shifts to a shorter wavelength while ω^- red shifts to a longer wavelength as the amount of Au precursor increases. As the amount of Au precursor increases,

the dipole moment (proportional to the radius of the Au particle) of Au satellites becomes larger, which significantly enhances the modal coupling between the cavity mode and the LSPR mode. Thus, the enhanced modal coupling leads to the blueshift of ω^+ and the redshift of ω^- with a larger modal splitting in extinction spectra. The amount of HAuCl₄ is found to play a crucial role in tuning the modal coupling in the ACA structure because it affects not only the dipole of the Au satellites but also the thickness of the Cu₂O shell in between. Therefore, an increase in the peak intensities of ω^+ and ω^- modes was observed due to the enhanced modal coupling as the amount of HAuCl₄ increased at first with well-defined cavity mode coupling with the dipole modes of Au satellites. However, the peak intensity started to decrease when the amount of HAuCl₄ was in excess, leading to the overgrowth of Au satellites and the ill-defined Cu₂O shell, *i.e.*, the less-defined nanocavity mode with weaker modal coupling (Fig. 3A and B). Therefore, the dual-extinction bands can be observed in Fig. 3B while not obvious in Fig. 3A because of the different degrees of separation between the two modes. However, the dual-extinction bands can't be observed in the case where the Cu₂O shell is very thick (Fig. S12D†). We further compared the full width at half maximum (FWHM) and the peak area changes in both cases (Fig. 3C). Both FWHM and peak area follow the trend of first increasing and then decreasing (Fig. 3C). To further understand the results of FWHM and peak area with the satellite variation, we compared the changes in FWHM and peak area for ACA NPs with 5 different Cu₂O thicknesses (Fig. S13†). All the samples showed a consistent variation trend with an increase first and a decrease later as the amount of HAuCl₄ increased. In particular, sample S13D with proper Cu₂O shell thickness represents the optimum condition for the modal coupling, which was preserved even with a higher amount of HAuCl₄. Therefore, our

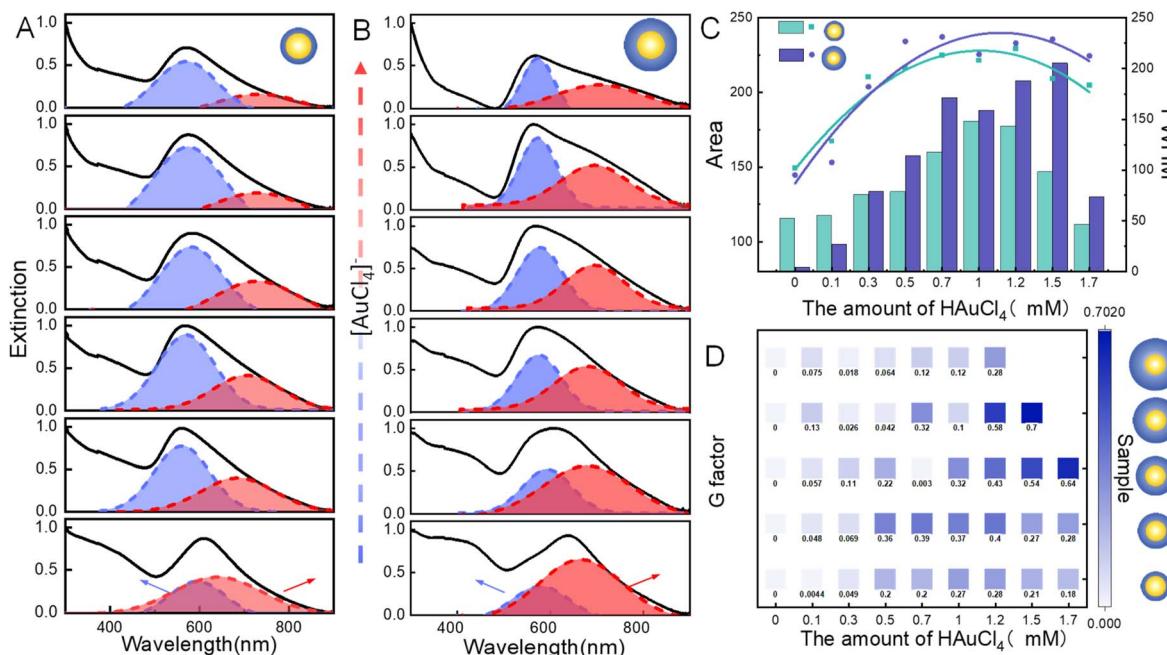


Fig. 3 Tuning the modal splitting in the extinction spectra of ACA NPs with different Cu_2O thicknesses. (A and B) Extinction spectra of ACA NPs with different Cu_2O thicknesses (the dipole resonance peak located at 630 nm and 667 nm, respectively) as a function of the amount of Au precursor. The extinction spectra were fitted using the Gaussian function. The red and blue curves indicate the Gaussian fitting of the dual extinction bands. The concentration of HAuCl_4 varies from 0.3, 0.5, 1.0, 1.2, 1.5 to 1.7 mM (bottom to top) in panels (A and B). (C) Histogram of FWHM and peak area (extracted from the extinction spectra shown in panels (A and B)) as a function of the amount of Au precursor. (D) The G_{factor} of ACA NPs with different Cu_2O shell thicknesses as a function of the amount of Au precursor. The G_{factor} was color-coded with different shades and shown with corresponding values. The thickness of the Cu_2O increases from bottom to top.

experimental observations suggested that the shell thickness of the Cu_2O layer plays a crucial role in modulating the modal splitting of optical extinction spectra.

To better understand the effect of Au satellites and Cu_2O shells on the optical properties of ACA NPs, the Mie theory was employed for a deeper analysis. We consider $\text{Au}@\text{Cu}_2\text{O}$ NPs as the LSPR core coated with a dielectric shell, which is also in line with previous studies.^{26,29,46,47} For the $\text{Au}@\text{Cu}_2\text{O}$ NPs, an individual dipole resonance peak in the ensemble extinction spectra (Fig. 1F) was observed instead of a complicated multipolar optical mode,^{48,49} which is also in excellent agreement with our simulation results (Fig. 2B). Also, it is difficult to simply consider it as a Mie resonator due to the presence of the Au core in $\text{Au}@\text{Cu}_2\text{O}$ NPs which significantly changes its optical response.⁵⁰ For NPs much smaller than the wavelength of the absorbing light, only the dipole term is assumed to contribute to the absorption, while quadrupole resonances and other high-order modes are neglected.⁵¹ Thus, as depicted by Christian, both the cavity and the orbit of satellites independently form purely dipolar polarization states.⁵² It has been reported that a strong red shift and spectral peak broadening occur when the Au spherical monomer forms the dimer. Therefore, the spectral line shapes of extinction spectra reflect the interaction between two plasmonic metal NPs to a certain extent. The simplest description of this interaction, the coupling of two nearby oscillators, is in the coupling of two nearby dipoles. The interaction energy is given by⁵³

$$V \propto \frac{P_1 P_2}{r^3} \quad (1)$$

where P_1 and P_2 are the magnitudes of the dipole moments and r is the interparticle distance. The dipole moments of $\text{Au}@\text{Cu}_2\text{O}$ NPs can be calculated from the Mie theory described by⁵⁴

$$P_{\text{cavity}} = \frac{\varepsilon_{\text{metal}} - \varepsilon_{\text{dielectric}}}{\varepsilon_{\text{metal}} + 2\varepsilon_{\text{dielectric}}} a^3 E_0 \quad (2)$$

where $\varepsilon_{\text{metal}}$ is the dielectric constant of the metal core, $\varepsilon_{\text{dielectric}}$ is the dielectric constant of the dielectric shell, a is the metal NP radius and E_0 refers to the electromagnetic (EM) field composed of the incident radiation. Since the thickness of the dielectric Cu_2O shell in $\text{Au}@\text{Cu}_2\text{O}$ NPs also plays a crucial role in affecting the optical properties, we further included this term in eqn (2), which is extensively discussed in the ESI† as the additional notes.

To simplify the ACA system, we make the following transformations analogous to self-consistent field theory in solving multi-electron atomic wave functions. If we separately consider the individual components in the coupled nanostructures, the EM field distribution and optical performance of one component will be influenced by the induced interaction of dipolar resonances with other nearby components. Under the incident light, the dipole moment of a metal NP is proportional to the size of the particle, while the vector sum of dipole moments of the cavity and other satellites can be described as follows:

$$\vec{P}_2 = \vec{P}_{\text{cavity}} + \sum_{i=1}^N \vec{P}_i \quad (3)$$

where P_i is the dipole moment of satellite i . Thus, the model is simplified as the interaction between a dipole (one satellite in the system) and another dipole (system consisting of other satellites with cavity), and the interaction can be described using eqn (1), which is similar to the coupling in an asymmetric Au NP dimer.⁵⁵ The term r^3 in eqn (1) shows that the interaction decays rapidly when moving away, explaining the experimental phenomenon well that there are few changes in the extinction spectrum of sample 5 (Fig. S12D†). This model qualitatively explains well why the size of the Au satellites and the thickness of the Cu₂O shell have an impact on the coupling interaction and account for the changes in the extinction spectra of coupled nanostructures with smaller satellites in previous work,³⁶ attributed to weak interaction energy. When we increase the amount of HAuCl₄, the diameter of Au satellites increases, which is reflected in the increase of the linewidth of extinction spectra, implying stronger interaction between coupled harmonic oscillators.

To further quantify the spectral change before and after loading Au satellites on ACA NPs, we defined an equation described by the ratio of the skewness ($\Delta\lambda$, Fig. S14†) of the peak to the full peak width as follows:

$$G_{\text{factor}} = \frac{\Delta\lambda}{2\text{FWHM}} \quad (4)$$

We compared the G_{factor} of ACA NPs with different Cu₂O shell thicknesses as a function of the amount of Au precursor. The G_{factor} was color-coded with different shades and shown with corresponding values. As shown in Fig. 3D, the value of the G_{factor} follows a volcano curve as the amount of HAuCl₄ increases for ACA NPs with thinner Cu₂O shells. In contrast, the G_{factor} value of ACA NPs with thicker Cu₂O shells gradually increases as the amount of Au precursors increases, which could be attributed to the dominance of the ω^- mode. Note that there are no specific physical meanings of the G value as we designed, so we are not able to claim whether the system satisfies the strong coupling regime or not based on the G values.

To more quantitatively understand the impact of plasmon coupling on the optical properties of the ACA hybrid structure, we further carried out the single-particle DFS measurements on the ACA NPs (Fig. 4). The scattering spectra of ~30 randomly selected particles for each sample were measured using a hyperspectral detection system (Fig. S15,† see experimental details in the ESI†).⁵⁶ We used the Au@Cu₂O NPs whose scattering peak is located at 650 nm as a model system to investigate the spectral evolution of scattering spectra before and after

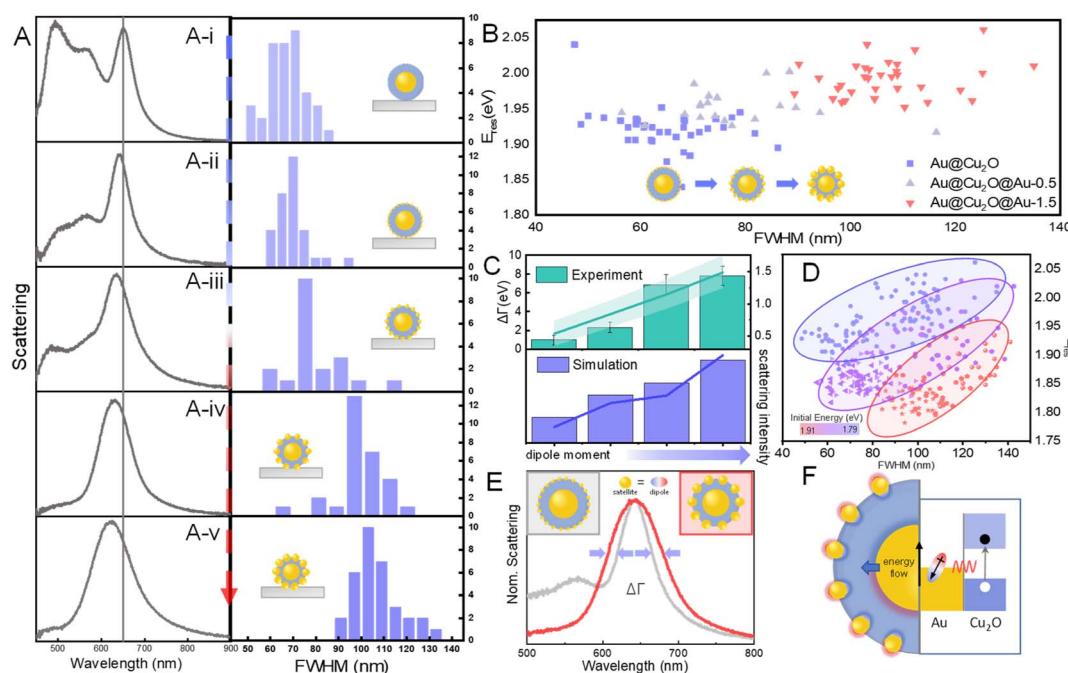


Fig. 4 Tuning the modal coupling in the scattering spectra of ACA NPs at the single-particle level. (A) Single-particle scattering spectra of ACA NPs loading Au satellites of different sizes and the corresponding statistical histogram of FWHM. The solid line shows the peak position of the scattering spectrum of Au@Cu₂O NPs. The insets show the corresponding geometric models of ACA NPs on the ITO substrate. (B) Γ vs. E_{res} of ACA NPs with different sizes of Au satellites. E_{res} is plotted as a function of Γ (FWHM) and color-coded by different samples (inset) from ~30 individual particles, respectively. (C) Comparison of experimental and simulated changes in intensity and FWHM of scattering spectra with increasing dipole moment of Au satellites. (D) Histogram showing the correlation between Γ and E_{res} for the scattering spectra of ACA NPs with varying sizes of Au satellites. Each color-coded elliptical region represents the change in Γ with E_{res} of ACA NPs with varying sizes of dipole. (E) Schematic illustration of the plasmon damping of the ACA structure with varying sizes of dipole. (F) Schematic illustration of the energy transfer mechanism in the coupled structure of ACA NPs.



forming Au satellites with different sizes and densities (Fig. 4A). The dipole scattering peak at \sim 650 nm of Au@Cu₂O NPs shows a slight blueshift as the amount of Au precursors increases. More importantly, the scattering peak becomes significantly broadening along with increases in FWHM, as revealed by the statistical histogram of each dataset. As shown in Fig. 4B, we further correlated the resonance energy (E_{res}) of the scattering peak with the FWHM (Γ). Again, it is found that the E_{res} increases along with the increases in FWHM as the size of Au satellites increases, which is also in excellent agreement with the simulated results (Fig. 4C). Note that the thickness of the Cu₂O shell decreases as more Au precursors react with the shell through the GRR, which in turn increases the coupling strength that leads to the increase in FWHM. We also designed and fabricated Au@Cu₂O NPs with two different shell thicknesses to tune the plasmon coupling when forming ACA NPs with different thicknesses of Cu₂O cavities (Fig. S16 and S17†). We then classified them into coupled structures with different initial energies of 1.91 eV (blue), 1.85 eV (purple), and 1.79 eV (red), as shown in Fig. 4D. It is worth noting that as we added HAuCl₄ to the system, it is difficult to ensure that each particle undergoes the same changes. However, single-particle spectroscopy has an advantage over ensemble extinction spectroscopy, which allows one to count the changes in each particle. Thus, we counted all the samples with different HAuCl₄ added at the same initial Cu₂O thickness, and the trend of FWHM and peak position at that thickness can be obtained from the scatter plot (Fig. 4D). By comparing the scatter plots of Au@Cu₂O NPs with three different initial energies, we found that the resonance energy of Au@Cu₂O with higher initial energy varies faster with the FWHM. While the modal peak splitting cannot be observed in the single-particle scattering spectra, the dramatic broadening of scattering plasmonic modes also indicates the enhanced coupling between the plasmon mode of the Au satellite and the nanocavity mode of the inner Au@Cu₂O layer.

We further compared the FWHM of scattering spectra of ACA NPs with significantly different sizes of Au satellites (Fig. 4E). It is observed that the larger Au satellite induces significantly more plasmon damping, resulting in the broadening of the scattering peak and an increase in FWHM. The changes in FWHM could be attributed to the enhanced electron-phonon relaxation when surface adsorbates induce an electric dipole at the particle surfaces,⁵⁷ which has also been previously demonstrated using single-particle spectroscopy.⁵⁸ Recent studies have further demonstrated that energy transfer can cause a decrease in the electron temperature, which in turn decreases the electron-phonon relaxation time.⁵⁹ In the model of the ACA system, Au satellites serving as dipoles can accelerate the electron-phonon relaxation of the Au core, which accelerates the energy transfer process. The electron-phonon coupling time ($\tau_{\text{e-ph}}$) is determined by the electronic temperature. The dependence of $\tau_{\text{e-ph}}$ can be described using the following equation,⁶⁰ according to the well-established two-temperature model for describing the electron dynamics of metal NPs.

$$\tau_{\text{e-ph}} \approx \frac{\gamma T_1}{g} + \frac{(1 - P_i)U}{2gT_1} \quad (5)$$

where γ is the electron heat capacity coefficient of Au, g is the electron-phonon coupling constant, T_1 is the lattice temperature, U is the initial absorbed energy after SPR excitation, and P_i is the fraction of energy injected into Cu₂O. Based on the results and discussion above, a possible mechanism for plasmon-induced energy transfer is proposed in Fig. 4F. Au satellites act as dipoles that accelerate the energy flow from the Au core to the Cu₂O shell. Note that we only consider the contribution of Au satellites as electric dipoles and ignore their contribution as catalytically active sites in this case. The Au satellites could also serve as catalytically active sites toward plasmon-driven photocatalytic reactions, which can be verified by following *in situ* SERS measurements.

To further explore the significance of the Au core on the enhanced coupled plasmonic properties of ACA NPs, we carried out the plasmon-driven dimerization of 4-nitrothiophenol (4-NTP) to 4,4'-dimercaptoazobenzene (DMAB) on ACA NPs; in this case, the Au satellites serve as the catalytically active sites (Fig. 5). Time-resolved SERS was used as a spectroscopic tool to monitor the molecular transformation from 4-NTP to DMAB upon laser excitation, in which the characteristic SERS peaks of 4-NTP and DMAB have been carefully demonstrated by previous studies (Fig. S18†).^{61,62} Similar to plasmonic NPs on mirror systems, the Au core is expected to serve as a plasmonic mirror to confine the electromagnetic field of light into a dielectric layer and significantly enhance the plasmonic near-fields.⁶³⁻⁶⁷ As illustrated in Fig. 5A, a CW excitation laser was focused on a 2 μm diameter focal spot by using a confocal Raman microscope. ACA NPs and Cu₂O@Au NPs with roughly similar sizes and densities of Au satellites were employed as the single-particle bifunctional substrates (Fig. S19†). Fig. 5B and C show the temporal evolution of SERS spectral features under isolated air at an excitation wavelength (λ_{ex}) of 785 nm on ACA NPs and Cu₂O NPs, respectively. Since the tests were performed in the presence of water, we excluded photothermal effects on the reaction and only considered carrier effects. At the initial stage, three SERS peaks located at 1078, 1338, and 1572 cm^{-1} are observed in the Raman spectrum, corresponding to the characteristic C-S bond stretching ($\nu_{\text{C-S}}$), symmetric nitro stretching (ν_{NO_2}) and aromatic ring stretching (ν_{ring}) modes of 4-NTP.⁶⁸⁻⁷⁰ After the laser excitation, four new SERS bands signifying the formation of DMAB emerged at 1143, 1181, 1444, and 1472 cm^{-1} , all of which became progressively more intense as the dimerization reaction proceeded.⁷⁰⁻⁷² The overall trend of 4-NTP on Cu₂O@Au follows the same law, despite some details different from ACA NPs, which is attributed to the different coupling rates (Fig. S19 and S20†). We further calculated the apparent fractions of DMAB and 4-NTP, denoted as θ_{DMAB} and θ_{NTP} , respectively, based on the relative intensities of the SERS peaks at 1438 cm^{-1} ($\nu_{\text{N=N}}$ of DMAB) and 1338 cm^{-1} (ν_{NO_2} of 4-NTP) with respect to the ν_{ring} mode at 1572 cm^{-1} (see details in the ESI†). As shown in Fig. 5D and E, the temporal evolutions of θ_{DMAB} and θ_{NTP} both obey an apparent first-order rate law, which can be described using the following rate equations:



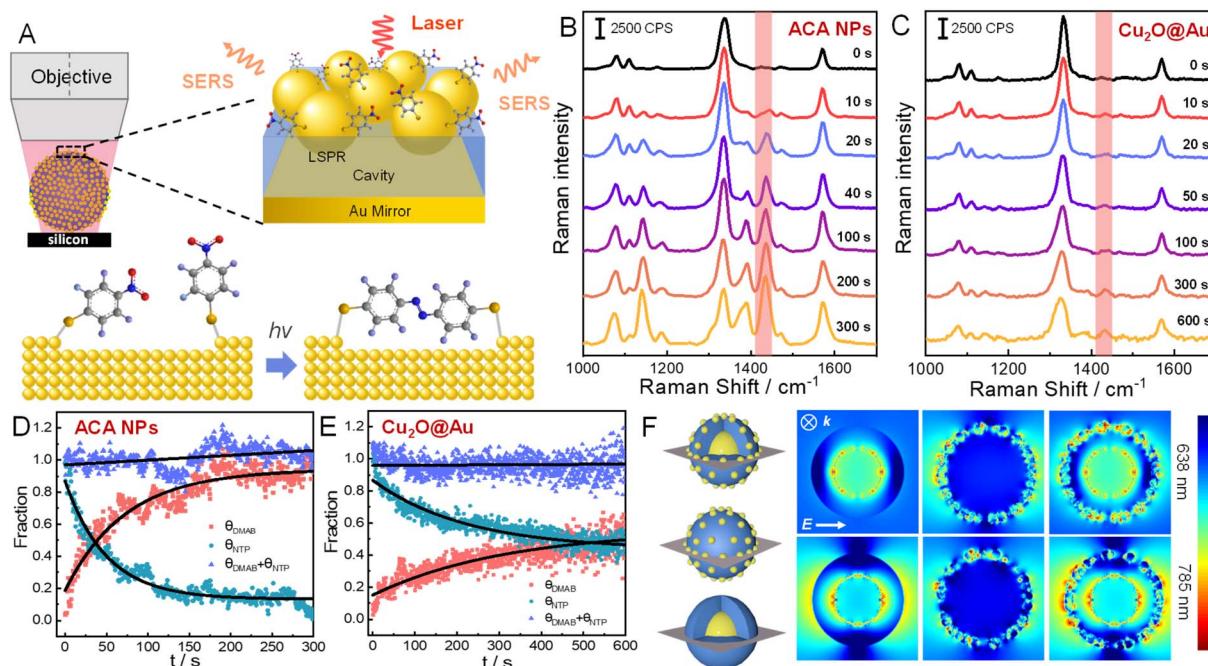


Fig. 5 (A) Schematic illustration of using SERS to monitor the plasmon-driven dimerization of 4-nitrothiophenol (4-NTP) on ACA NPs. (B and C) Comparison of time-resolved SERS spectra of 4-NTP on (B) ACA and (C) Cu₂O@Au NPs upon 785 nm laser excitation. (D and E) Temporal evolutions of θ_{NTP} , θ_{DMAB} , and $\theta_{\text{DMAB}} + \theta_{\text{NTP}}$ on (D) ACA and (E) Cu₂O@Au NPs. The solid lines were obtained by performing least-squares curve fitting to the experimental results. (F) Cross-sectional views of calculated near-field enhancements of Au@Cu₂O, Cu₂O@Au, and ACA NPs (left to right) at 638 nm and 785 nm excitation, respectively. The field enhancements are plotted on a logarithmic scale ($\log|E/E_0|^2$). E: electric field; k: incident wave vector.

$$\theta_{\text{DMAB}}(t) = \theta_{t=\infty}(t) \times (1 - e^{-kt}) \quad (6)$$

and

$$\theta_{\text{NTP}}(t) = 1 - \theta_{t=\infty} + \theta_{t=\infty} \times e^{-kt} \quad (7)$$

where k is the apparent first-order rate constant and $\theta_{t=\infty}$ is the maximum yield of DMAB achievable at infinitely long reaction time. The three key parameters could be obtained by performing least-squares curve fitting to the experimental results. The rate constant for the 4-NTP coupling reaction on ACA NPs is significantly higher than that on Cu₂O@Au NPs. Moreover, the size and density of Au satellites in ACA NPs have also been optimized at an optimal amount of Au precursors for maximizing the rate constant (Fig. S21†). Either increasing or decreasing the amount of Au precursor results in a smaller rate constant in comparison to the results shown in Fig. 2A. These results suggested that the presence of the Au core in ACA NPs significantly enhanced the efficiency of plasmon-induced hot carriers for photochemical molecular transformations.

Strong electromagnetic field enhancement can promote hot carrier separation more efficiently in metal-semiconductor nanostructures for photocatalysis, which has been widely studied.^{73–75} We propose that the different reaction behaviors between ACA NPs and Cu₂O@Au NPs in plasmon-driven dimerization of 4-NTP could be attributed to the different excitation strengths of plasmonic near-fields. To gain deep insights into the plasmonic near-field properties of different

structures, we have further used FDTD to calculate the near-field enhancement of ACA, Cu₂O@Au, and Au@Cu₂O NPs (Fig. 5F). Fig. 5F shows the cross-sectional views of calculated near-field enhancements of ACA, Cu₂O@Au, and Au@Cu₂O NPs ($|E/E_0|^2$ plotted on a logarithmic scale) under 638 nm and 785 nm laser excitation, respectively. Near-field enhancement of Au@Cu₂O NPs is mainly concentrated at the interface between the core and shell layer while weak on the outer surface of NPs. For Cu₂O@Au NPs, the external electric field of NPs is excited, while the internal near-field enhancement is weak. The enhanced electric field in Fig. 5F (center: Cu₂O@Au NPs) is due to the presence of Au satellites that create inter-particle hot-spots on the surface of Cu₂O NPs. In contrast, ACA NPs exhibit an enormous enhancement of both external and internal near-fields, and this synergistic effect greatly enhances the overall electromagnetic field for plasmonic photochemistry.

We further employed ultrafast TA spectroscopy to investigate the dynamics of plasmon decay of ACA hybrid structures in comparison to the corresponding Au@Cu₂O NPs (Fig. 6A). Both TA spectra of ACA and Au@Cu₂O NPs exhibit a bleaching signal at the plasmon resonance accompanied by two positive wings on both sides due to induced absorption (Fig. 6B and C). The electron–electron scattering occurred during the initial fast rise due to the high electron density of metals (Fig. 6D). In addition, a longer absorption rise can be observed for ACA NPs compared with Au and Au@Cu₂O NPs (Fig. S22–S24†), which is consistent with previous observations under coupling conditions.⁶⁶ The ultrafast decay arises from electron–phonon scattering on the

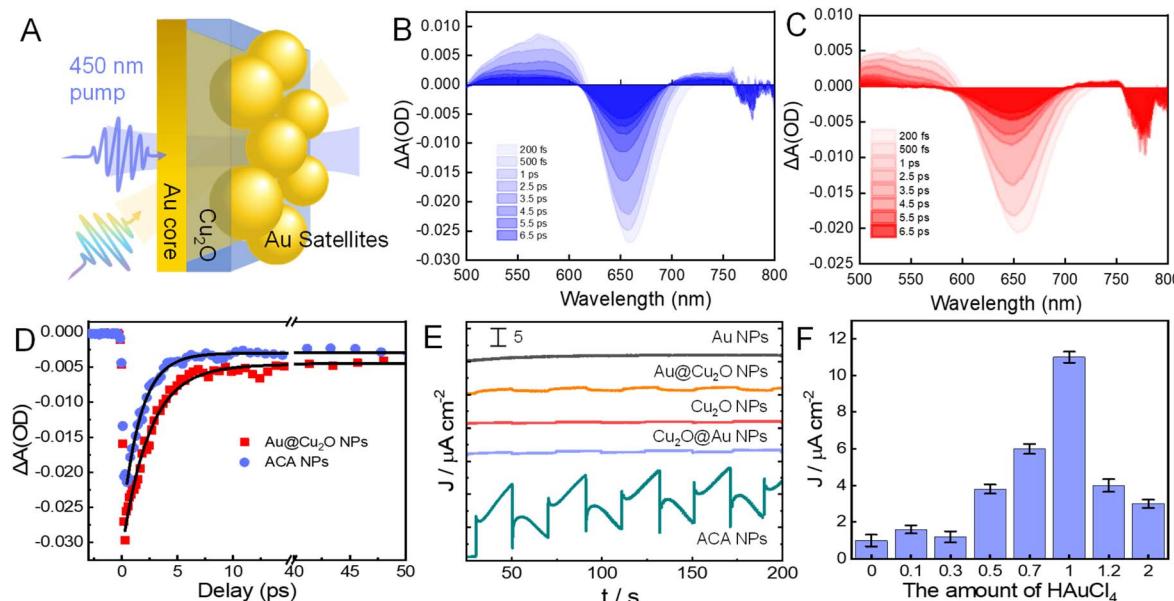


Fig. 6 (A) Schematic illustration of the ultrafast transient absorption (TA) spectroscopic measurements on ACA NPs. (B) and (C) TA spectra of (B) Au@Cu₂O and (C) ACA NPs, respectively, with various delay times under a 450 nm pump pulse. (D) Kinetics of the photoinduced bleach signal at 660 nm for Au@Cu₂O and ACA NPs. The results of the least-squares fitting are shown as solid curves. (E) Plots of photocurrent as a function of time for Au RCO, Au@Cu₂O, and ACA NPs according to light on/off at the same cathode applied potential (visible light illumination: $\lambda > 400$ nm). (F) Histogram of photocurrents of ACA NPs under different coupling conditions tuned by changing the amount of Au precursors.

order of a few picoseconds followed by the lattice cooling at a longer time scale. The fitted data revealed electron–phonon relaxation times ($\tau_{e\text{-}ph}$) of 2.45 and 1.65 ps for Au@Cu₂O and ACA NPs (Table S1†), which is consistent with our inference that the Au satellites on the surface of the ACA NPs act as dipoles, accelerating the electron–phonon relaxation process of the Au core. In contrast, the Au NPs exhibited a long $\tau_{e\text{-}ph}$ (4.89 ps), which is in line with previous studies.

To investigate the photocatalytic activities of the as-constructed ACA hybrid structures, we measured the photocurrents of ACA NPs in a three-electrode system using a Xe lamp as the light source (see details in the ESI†). Notably, charge recombination is significantly suppressed in PEC measurement with an applied bias voltage, allowing us to focus on how plasmonic charge density affects photocatalytic reactions. Fig. 6E shows the comparison of the photocurrent response among the Au, Au@Cu₂O, Cu₂O, Cu₂O@Au, and ACA NPs. No significant photocurrent response is observed in the Au NP, whereas a stronger photocurrent response occurs in Au@Cu₂O NPs. Remarkably, at an applied potential of -0.3 V, the photocurrent density of ACA NPs is 5-fold with respect to that of Au@Cu₂O NPs. Additionally, Cu₂O and Cu₂O@Au NPs show similar photocurrents to the Au@Cu₂O NPs, suggesting that the formation of ACA NPs plays a crucial role in the enhanced photocurrents. As shown in Fig. 6F, we further compared the photocurrent response of ACA NPs with different densities and sizes of Au satellites (samples as shown in Fig. 2A). First, all the ACA NPs exhibit significantly higher photocurrent responses compared to the Au@Cu₂O NPs. Moreover, the photocurrent responses of ACA follow the same trend of increasing and then

decreasing (Fig. 2F and S25†), consistent with our analysis of changes in the line width and peak area (Fig. 3C) as well as previous work on photocatalytic hydrogenation.³⁷ While the correlation is very intriguing, the underlying mechanism could be much more complicated than we expected and require further scrutiny. Overall, our experimental observations strongly demonstrate the superior performance of the ACA hybrid structure toward plasmonic photocatalysis.

Finally, the method for constructing the core–shell–satellite structure that we developed here also shows versatility for creating unconventional hybrid plasmon structures for various optical applications (Fig. S26 and S27†). For instance, by using a chiral Au rhombic dodecahedron (RD) as the initial core,⁷⁶ we were able to demonstrate the plasmon coupling between the Au satellite and chiral nanocavity through the construction of chiral Au RD@Cu₂O@Au hybrid structures (Fig. S26†).⁷⁷ By simply controlling the amount of Au precursors that react with the Cu₂O shell through the GRR, the chiral ACA NPs with tunable chiral couplings and chiroptical activities can be obtained (Fig. S26 and S27†). Moreover, significantly enhanced electromagnetic fields were observed for chiral ACA structures because of the intense chiral plasmon coupling in the core–shell–satellite structure (Fig. S27†), and this phenomenon has also been observed in the mentioned achiral ACA NPs (Fig. 5F). FDTD simulations further revealed the chiral near-field distributions when excited using 824 nm left/right-handed circularly polarized (LCP/RCP) light (Fig. S27†). It is envisioned that the chiral ACA structure with significantly enhanced plasmon coupling and chiral near-fields could provide a promising platform for chirality-dependent plasmonic photocatalysis.^{78–82}



Conclusions

In summary, we have demonstrated a general approach for constructing core–shell–satellite coupled nanostructures with tunable optical properties for enhanced plasmon photocatalysis. By deliberately changing the size of Au satellites and the shell thickness of Cu_2O , we experimentally and theoretically demonstrated the modal splitting in the extinction spectra of the hybrid structures. We further employed single-particle scattering to show the spectral changes of plasmon bands on each particle. Moreover, the coupled nanostructures with tunable plasmon coupling exhibit excellent photocatalytic activities toward plasmon-driven chemical reactions, as revealed by *in situ* SERS measurements. TA spectroscopy and PEC measurements also show the superior performance of the hybrid structure toward plasmonic photocatalysis. Finally, this strategy can also be employed to construct chiral coupled nanostructures for applications in chirality-dependent plasmonic photocatalysis. It is envisioned that integrating plasmon coupling and excellent catalytic activity within the 3D coupled nanostructure represents a promising approach for applications in photonics and plasmonic photocatalysis.

Data availability

A data availability statement (DAS) is required to be submitted alongside all articles. Please read our full guidance on data availability statements for more details and examples of suitable statements you can use.

Author contributions

The manuscript was written through the contributions of all authors. All authors have given approval to the final version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (Grant No. 22174104, 22472119, and 22405195). This work was also funded by the Basic Research Program of Jiangsu (Grant No. BK20240448 to L. S.). The authors also acknowledge the support of the Large-scale Instrument and Equipment Sharing Foundation of Wuhan University and the Core Facility of Wuhan University. We also thank the Core Research Facility of the College of Chemistry and Molecular Sciences at Wuhan University for SEM and TEM characterization studies.

Notes and references

- 1 W. L. Barnes, A. Dereux and T. W. Ebbesen, Surface plasmon subwavelength optics, *Nature*, 2003, **424**, 824–830.

- 2 F. J. García de Abajo, Nonlocal Effects in the Plasmons of Strongly Interacting Nanoparticles, Dimers, and Waveguides, *J. Phys. Chem. C*, 2008, **112**, 17983–17987.
- 3 P. K. Jain, S. Eustis and M. A. El-Sayed, Plasmon Coupling in Nanorod Assemblies: Optical Absorption, Discrete Dipole Approximation Simulation, and Exciton-Coupling Model, *J. Phys. Chem. B*, 2006, **110**, 18243–18253.
- 4 E. Prodan, C. Radloff, N. J. Halas and P. Nordlander, A Hybridization Model for the Plasmon Response of Complex Nanostructures, *Science*, 2003, **302**, 419–422.
- 5 Y. Bao, Z. Hu, Z. Li, X. Zhu and Z. Fang, Magnetic Plasmonic Fano Resonance at Optical Frequency, *Small*, 2015, **11**, 2177–2181.
- 6 N. Verellen, Y. Sonnefraud, H. Sobhani, F. Hao, V. V. Moshchalkov, P. V. Dorpe, P. Nordlander and S. A. Maier, Fano Resonances in Individual Coherent Plasmonic Nanocavities, *Nano Lett.*, 2009, **9**, 1663–1667.
- 7 S. Zhang, K. Bao, N. J. Halas, H. Xu and P. Nordlander, Substrate-Induced Fano Resonances of a Plasmonic Nanocube: A Route to Increased-Sensitivity Localized Surface Plasmon Resonance Sensors Revealed, *Nano Lett.*, 2011, **11**, 1657–1663.
- 8 N. Liu, L. Langguth, T. Weiss, J. Kästel, M. Fleischhauer, T. Pfau and H. Giessen, Plasmonic analogue of electromagnetically induced transparency at the Drude damping limit, *Nat. Mater.*, 2009, **8**, 758–762.
- 9 P. Zeng, J. Cadusch, D. Chakraborty, T. A. Smith, A. Roberts, J. E. Sader, T. J. Davis and D. E. Gómez, Photoinduced Electron Transfer in the Strong Coupling Regime: Waveguide–Plasmon Polaritons, *Nano Lett.*, 2016, **16**, 2651–2656.
- 10 N. S. King, L. Liu, X. Yang, B. Cerjan, H. O. Everitt, P. Nordlander and N. J. Halas, Fano Resonant Aluminum Nanoclusters for Plasmonic Colorimetric Sensing, *ACS Nano*, 2015, **9**, 10628–10636.
- 11 A. Aubry, D. Y. Lei, A. I. Fernández-Domínguez, Y. Sonnefraud, S. A. Maier and J. B. Pendry, Plasmonic Light-Harvesting Devices over the Whole Visible Spectrum, *Nano Lett.*, 2010, **10**, 2574–2579.
- 12 D. Liu and C. Xue, Plasmonic Coupling Architectures for Enhanced Photocatalysis, *Adv. Mater.*, 2021, **33**, 2005738.
- 13 S. Kawata, Y. Inouye and P. Verma, Plasmonics for near-field nano-imaging and superlensing, *Nat. Photonics*, 2009, **3**, 388–394.
- 14 H. Baida, D. Mongin, D. Christofilos, G. Bachelier, A. Crut, P. Maioli, N. Del Fatti and F. Vallée, Ultrafast Nonlinear Optical Response of a Single Gold Nanorod near Its Surface Plasmon Resonance, *Phys. Rev. Lett.*, 2011, **107**, 057402.
- 15 X. Hu, Y. Zhang, Y. Fu, H. Yang and Q. Gong, Low-Power and Ultrafast All-Optical Tunable Nanometer-Scale Photonic Metamaterials, *Adv. Mater.*, 2011, **23**, 4295–4300.
- 16 T. Oshikiri, K. Ueno and H. Misawa, Plasmon-Induced Ammonia Synthesis through Nitrogen Photofixation with Visible Light Irradiation, *Angew. Chem., Int. Ed.*, 2014, **53**, 9802–9805.



- 17 S. Li, P. Miao, Y. Zhang, J. Wu, B. Zhang, Y. Du, X. Han, J. Sun and P. Xu, Recent Advances in Plasmonic Nanostructures for Enhanced Photocatalysis and Electrocatalysis, *Adv. Mater.*, 2021, **33**, 2000086.
- 18 L. A. Zhou, J. M. P. Martirez, J. Finzel, C. Zhang, D. F. Swearer, S. Tian, H. Robatjazi, M. H. Lou, L. L. Dong, L. Henderson, P. Christopher, E. A. Carter, P. Nordlander and N. J. Halas, Light-driven methane dry reforming with single atomic site antenna-reactor plasmonic photocatalysts, *Nat. Energy*, 2020, **5**, 61–70.
- 19 T. Kawakaki, T. Nakagawa, M. Sakamoto and T. Teranishi, Carrier-Selective Blocking Layer Synergistically Improves the Plasmonic Enhancement Effect, *J. Am. Chem. Soc.*, 2019, **141**, 8402–8406.
- 20 M. Cai, Z. Wu, Z. Li, L. Wang, W. Sun, A. A. Tountas, C. Li, S. Wang, K. Feng, A.-B. Xu, S. Tang, A. Tavasoli, M. Peng, W. Liu, A. S. Helmy, L. He, G. A. Ozin and X. Zhang, Greenhouse-inspired supra-photothermal CO₂ catalysis, *Nat. Energy*, 2021, **6**, 807–814.
- 21 C. Clavero, Plasmon-induced hot-electron generation at nanoparticle/metal-oxide interfaces for photovoltaic and photocatalytic devices, *Nat. Photonics*, 2014, **8**, 95–103.
- 22 S. K. Cushing, J. Li, F. Meng, T. R. Senty, S. Suri, M. Zhi, M. Li, A. D. Bristow and N. Wu, Photocatalytic Activity Enhanced by Plasmonic Resonant Energy Transfer from Metal to Semiconductor, *J. Am. Chem. Soc.*, 2012, **134**, 15033–15041.
- 23 R. Long, K. Mao, M. Gong, S. Zhou, J. Hu, M. Zhi, Y. You, S. Bai, J. Jiang, Q. Zhang, X. Wu and Y. Xiong, Tunable Oxygen Activation for Catalytic Organic Oxidation: Schottky Junction versus Plasmonic Effects, *Angew. Chem., Int. Ed.*, 2014, **53**, 3205–3209.
- 24 K. Song, H. Lee, M. Lee and J. Y. Park, Plasmonic Hot Hole-Driven Water Splitting on Au Nanoprism/P-Type GaN, *ACS Energy Lett.*, 2021, **6**, 1333–1339.
- 25 J. W. Hong, D. H. Wi, S.-U. Lee and S. W. Han, Metal–Semiconductor Heteronanoocrystals with Desired Configurations for Plasmonic Photocatalysis, *J. Am. Chem. Soc.*, 2016, **138**, 15766–15773.
- 26 H. Jia, F. Li, T. H. Chow, X. Liu, H. Zhang, Y. Lu, J. Wang and C.-y. Zhang, Construction of Spatially Separated Gold Nanocrystal/Cuprous Oxide Architecture for Plasmon-Driven CO₂ Reduction, *Nano Lett.*, 2022, **22**, 7268–7274.
- 27 S. Linic, S. Chavez and R. Elias, Flow and extraction of energy and charge carriers in hybrid plasmonic nanostructures, *Nat. Mater.*, 2021, **20**, 916–924.
- 28 R. Long, Z. Rao, K. Mao, Y. Li, C. Zhang, Q. Liu, C. Wang, Z.-Y. Li, X. Wu and Y. Xiong, Efficient Coupling of Solar Energy to Catalytic Hydrogenation by Using Well-Designed Palladium Nanostructures, *Angew. Chem., Int. Ed.*, 2015, **54**, 2425–2430.
- 29 W. Xu, J. Jia, T. Wang, C. Li, B. He, J. Zong, Y. Wang, H. J. Fan, H. Xu, Y. Feng and H. Chen, Continuous Tuning of Au–Cu₂O Janus Nanostructures for Efficient Charge Separation, *Angew. Chem., Int. Ed.*, 2020, **59**, 22246–22251.
- 30 J. Zhang, X. Jin, P. I. Morales-Guzman, X. Yu, H. Liu, H. Zhang, L. Razzari and J. P. Claverie, Engineering the Absorption and Field Enhancement Properties of Au–TiO₂ Nanohybrids via Whispering Gallery Mode Resonances for Photocatalytic Water Splitting, *ACS Nano*, 2016, **10**, 4496–4503.
- 31 P. Ding, E. Liang, G. Cai, W. Hu, C. Fan and Q. Xue, Dual-band perfect absorption and field enhancement by interaction between localized and propagating surface plasmons in optical metamaterials, *J. Opt.*, 2011, **13**, 075005.
- 32 J. Yang, Q. Sun, K. Ueno, X. Shi, T. Oshikiri, H. Misawa and Q. Gong, Manipulation of the dephasing time by strong coupling between localized and propagating surface plasmon modes, *Nat. Commun.*, 2018, **9**, 4858.
- 33 X. Shi, K. Ueno, T. Oshikiri, Q. Sun, K. Sasaki and H. Misawa, Enhanced water splitting under modal strong coupling conditions, *Nat. Nanotechnol.*, 2018, **13**, 953–958.
- 34 Y. Suganami, T. Oshikiri, X. Shi and H. Misawa, Water Oxidation under Modal Ultrastrong Coupling Conditions Using Gold/Silver Alloy Nanoparticles and Fabry-Pérot Nanocavities, *Angew. Chem., Int. Ed.*, 2021, **60**, 18438–18442.
- 35 Y. Cao, X. Shi, T. Oshikiri, S. Zu, Y. Sunaba, K. Sasaki and H. Misawa, Near-field engineering for boosting the photoelectrochemical activity to a modal strong coupling structure, *Chem. Commun.*, 2021, **57**, 524–527.
- 36 X. Yu, X. Liu, B. Wang, Q. Meng, S. Sun, Y. Tang and K. Zhao, An LSPR-based “push–pull” synergetic effect for the enhanced photocatalytic performance of a gold nanorod@cuprous oxide-gold nanoparticle ternary composite, *Nanoscale*, 2020, **12**, 1912–1920.
- 37 H. Ren, J.-L. Yang, W.-M. Yang, H.-L. Zhong, J.-S. Lin, P. M. Radjenovic, L. Sun, H. Zhang, J. Xu, Z.-Q. Tian and J.-F. Li, Core–Shell–Satellite Plasmonic Photocatalyst for Broad-Spectrum Photocatalytic Water Splitting, *ACS Mater. Lett.*, 2021, **3**, 69–76.
- 38 Y. Kim, D. H. Wi, J. W. Hong and S. W. Han, Plasmonic Nanocrystal Assembly–Semiconductor Hybrids for Boosting Visible to Near-Infrared Photocatalysis, *ACS Nano*, 2023, **17**, 18641–18651.
- 39 A. F. Kockum, A. Miranowicz, S. De Liberato, S. Savasta and F. Nori, Ultrastrong coupling between light and matter, *Nat. Rev. Phys.*, 2019, **1**, 19–40.
- 40 R. Houdre, J. L. Gibernon, P. Pellandini, R. P. Stanley, U. Oesterle, C. Weisbuch, J. Ogorman, B. Roycroft and M. Ilegems, Saturation of the Strong-Coupling Regime in a Semiconductor Microcavity - Free-Carrier Bleaching of Cavity Polaritons, *Phys. Rev. B*, 1995, **52**, 7810–7813.
- 41 T. H. Chow, Y. Lai, X. Cui, W. Lu, X. Zhuo and J. Wang, Colloidal Gold Nanorings and Their Plasmon Coupling with Gold Nanospheres, *Small*, 2019, **15**, 1902608.
- 42 W. Chen, S. Zhang, Q. Deng and H. Xu, Probing of sub-picometer vertical differential resolutions using cavity plasmons, *Nat. Commun.*, 2018, **9**, 801.
- 43 G.-C. Li, Y.-L. Zhang, J. Jiang, Y. Luo and D. Y. Lei, Metal–Substrate-Mediated Plasmon Hybridization in a Nanoparticle Dimer for Photoluminescence Line-Width Shrinking and Intensity Enhancement, *ACS Nano*, 2017, **11**, 3067–3080.



- 44 W. Q. Li, G. Wang, X. N. Zhang, H. P. Geng, J. L. Shen, L. S. Wang, J. Zhao, L. F. Xu, L. J. Zhang, Y. Q. Wu, R. Z. Tai and G. Chen, Geometrical and morphological optimizations of plasmonic nanoarrays for high-performance SERS detection, *Nanoscale*, 2015, **7**, 15487–15494.
- 45 H. J. Kim, S. H. Lee, A. A. Upadhye, I. Ro, M. I. Tejedor-Tejedor, M. A. Anderson, W. B. Kim and G. W. Huber, Plasmon-Enhanced Photoelectrochemical Water Splitting with Size-Controllable Gold Nanodot Arrays, *ACS Nano*, 2014, **8**, 10756–10765.
- 46 L. Zhang, D. A. Blom and H. Wang, Au-Cu₂O Core-Shell Nanoparticles: A Hybrid Metal-Semiconductor Heteronanostructure with Geometrically Tunable Optical Properties, *Chem. Mater.*, 2011, **23**, 4587–4598.
- 47 H. Jing, N. Large, Q. F. Zhang and H. Wang, Epitaxial Growth of Cu₂O on Ag Allows for Fine Control Over Particle Geometries and Optical Properties of Ag-Cu₂O Core-Shell Nanoparticles, *J. Phys. Chem. C*, 2014, **118**, 19948–19963.
- 48 T. Oshikiri, T. Hayakawa, H. Niinomi and M. Nakagawa, Strong Light Confinement by a Plasmon-Coupled Parabolic Nanoresonator Array, *J. Phys. Chem. C*, 2024, **128**, 5271–5279.
- 49 P. Gu, M. J. Wan, W. Y. Wu, Z. Chen and Z. L. Wang, Excitation and tuning of Fano-like cavity plasmon resonances in dielectric-metal core-shell resonators, *Nanoscale*, 2016, **8**, 10358–10363.
- 50 H. Sugimoto, T. Okazaki and M. Fujii, Mie Resonator Color Inks of Monodispersed and Perfectly Spherical Crystalline Silicon Nanoparticles, *Adv. Opt. Mater.*, 2020, **8**, 2000033.
- 51 S. Link and M. A. El-Sayed, Spectral Properties and Relaxation Dynamics of Surface Plasmon Electronic Oscillations in Gold and Silver Nanodots and Nanorods, *J. Phys. Chem. B*, 1999, **103**, 8410–8426.
- 52 R. P. M. Höller, M. Dulle, S. Thomä, M. Mayer, A. M. Steiner, S. Förster, A. Fery, C. Kuttner and M. Chanana, Protein-Assisted Assembly of Modular 3D Plasmonic Raspberry-like Core/Satellite Nanoclusters: Correlation of Structure and Optical Properties, *ACS Nano*, 2016, **10**, 5740–5750.
- 53 N. J. Halas, S. Lal, W.-S. Chang, S. Link and P. Nordlander, Plasmons in Strongly Coupled Metallic Nanostructures, *Chem. Rev.*, 2011, **111**, 3913–3961.
- 54 K. M. Kosuda, J. M. Bingham, K. L. Wustholz and R. P. Van Duyne, in *Comprehensive Nanoscience and Technology*, ed. D. L. Andrews, G. D. Scholes and G. P. Wiederrecht, Academic Press, Amsterdam, 2011, pp. 263–301.
- 55 S. Sheikholeslami, Y.-w. Jun, P. K. Jain and A. P. Alivisatos, Coupling of Optical Resonances in a Compositionally Asymmetric Plasmonic Nanoparticle Dimer, *Nano Lett.*, 2010, **10**, 2655–2660.
- 56 Q. F. Zhang, T. Hernandez, K. W. Smith, S. A. H. Jebeli, A. X. Dai, L. Warning, R. Baiyasi, L. A. McCarthy, H. Guo, D. H. Chen, J. A. Dionne, C. F. Landes and S. Link, Unraveling the origin of chirality from plasmonic nanoparticle-protein complexes, *Science*, 2019, **365**, 1475–1478.
- 57 S. L. Westcott, R. D. Averitt, J. A. Wolfgang, P. Nordlander and N. J. Halas, Adsorbate-Induced Quenching of Hot Electrons in Gold Core–Shell Nanoparticles, *J. Phys. Chem. B*, 2001, **105**, 9913–9917.
- 58 B. Foerster, V. A. Spata, E. A. Carter, C. Sönnichsen and S. Link, Plasmon damping depends on the chemical nature of the nanoparticle interface, *Sci. Adv.*, 2019, **5**, eaav0704.
- 59 G. Tagliabue, J. S. DuChene, M. Abdellah, A. Habib, D. J. Gosztola, Y. Hattori, W.-H. Cheng, K. Zheng, S. E. Canton, R. Sundararaman, J. Sá and H. A. Atwater, Ultrafast hot-hole injection modifies hot-electron dynamics in Au/p-GaN heterostructures, *Nat. Mater.*, 2020, **19**, 1312–1318.
- 60 D. C. Ratchford, A. D. Dunkelberger, I. Vurgaftman, J. C. Owrutsky and P. E. Pehrsson, Quantification of Efficient Plasmonic Hot-Electron Injection in Gold Nanoparticle-TiO₂ Films, *Nano Lett.*, 2017, **17**, 6047–6055.
- 61 Q. Zhang, Y. Zhou, X. Fu, E. Villarreal, L. Sun, S. Zou and H. Wang, Photothermal Effect, Local Field Dependence, and Charge Carrier Relaying Species in Plasmon-Driven Photocatalysis: A Case Study of Aerobic Nitrothiophenol Coupling Reaction, *J. Phys. Chem. C*, 2019, **123**, 26695–26704.
- 62 Y. Li, Z. Zhang, A. Du, X. Du, A. Zhu, C. Zhang, Y. Gao, Y. Hu, Y. Wang, X. Yang, L. Yang and W. Xie, Hot-Electron-Driven Interfacial Chemistry Using Non-Noble Plasmonic Cu under Visible-Light Irradiation, *ACS Photonics*, 2023, **10**, 3181–3187.
- 63 R. Chikkaraddy, V. A. Turek, N. Kongsuwan, F. Benz, C. Carnegie, T. van de Goor, B. de Nijs, A. Demetriadou, O. Hess, U. F. Keyser and J. J. Baumberg, Mapping Nanoscale Hotspots with Single-Molecule Emitters Assembled into Plasmonic Nanocavities Using DNA Origami, *Nano Lett.*, 2018, **18**, 405–411.
- 64 J.-H. Huh, J. Lee and S. Lee, Comparative Study of Plasmonic Resonances between the Roundest and Randomly Faceted Au Nanoparticles-on-Mirror Cavities, *ACS Photonics*, 2018, **5**, 413–421.
- 65 J. J. Baumberg, J. Aizpurua, M. H. Mikkelsen and D. R. Smith, Extreme nanophotonics from ultrathin metallic gaps, *Nat. Mater.*, 2019, **18**, 668–678.
- 66 Y. Gao, Q. Zhu, S. He, S. Wang, W. Nie, K. Wu, F. Fan and C. Li, Observation of Charge Separation Enhancement in Plasmonic Photocatalysts under Coupling Conditions, *Nano Lett.*, 2023, **23**, 3540–3548.
- 67 S. Hu, E. Elliott, A. Sánchez-Iglesias, J. Huang, C. Guo, Y. Hou, M. Kamp, E. S. A. Goerlitzer, K. Bedingfield, B. de Nijs, J. Peng, A. Demetriadou, L. M. Liz-Marzán and J. J. Baumberg, Full Control of Plasmonic Nanocavities Using Gold Decahedra-on-Mirror Constructs with Monodisperse Facets, *Adv. Sci.*, 2023, **10**, 2207178.
- 68 M. Sun and H. Xu, A Novel Application of Plasmonics: Plasmon-Driven Surface-Catalyzed Reactions, *Small*, 2012, **8**, 2777–2786.
- 69 Q. Zhang, Y. Zhou, E. Villarreal, Y. Lin, S. Zou and H. Wang, Faceted Gold Nanorods: Nanocuboids, Convex Nanocuboids, and Concave Nanocuboids, *Nano Lett.*, 2015, **15**, 4161–4169.



- 70 H.-K. Choi, K. S. Lee, H.-H. Shin and Z. H. Kim, Identification of the First Elementary Step in the Photocatalytic Reduction of Nitrobenzenethiols on a Metallic Surface, *J. Phys. Chem. Lett.*, 2016, **7**, 4099–4104.
- 71 Y.-F. Huang, H.-P. Zhu, G.-K. Liu, D.-Y. Wu, B. Ren and Z.-Q. Tian, When the Signal Is Not from the Original Molecule To Be Detected: Chemical Transformation of para-Aminothiophenol on Ag during the SERS Measurement, *J. Am. Chem. Soc.*, 2010, **132**, 9244–9246.
- 72 Q. Zhang and H. Wang, Mechanistic Insights on Plasmon-Driven Photocatalytic Oxidative Coupling of Thiophenol Derivatives: Evidence for Steady-State Photoactivated Oxygen, *J. Phys. Chem. C*, 2018, **122**, 5686–5697.
- 73 D. B. Ingram and S. Linic, Water Splitting on Composite Plasmonic-Metal/Semiconductor Photoelectrodes: Evidence for Selective Plasmon-Induced Formation of Charge Carriers near the Semiconductor Surface, *J. Am. Chem. Soc.*, 2011, **133**, 5202–5205.
- 74 Z. Liu, W. Hou, P. Pavaskar, M. Aykol and S. B. Cronin, Plasmon Resonant Enhancement of Photocatalytic Water Splitting Under Visible Illumination, *Nano Lett.*, 2011, **11**, 1111–1116.
- 75 H. Jin, M. Herran, E. Cortés and J. Lischner, Theory of Hot-Carrier Generation in Bimetallic Plasmonic Catalysts, *ACS Photonics*, 2023, **10**, 3629–3636.
- 76 X. Sun, J. Yang, L. Sun, G. Yang, C. Liu, Y. Tao, Q. Cheng, C. Wang, H. Xu and Q. Zhang, Tunable Reversal of Circular Dichroism in the Seed-Mediated Growth of Bichiral Plasmonic Nanoparticles, *ACS Nano*, 2022, **16**, 19174–19186.
- 77 G. Yang, Y. Tao, Q. Cheng, C. Liu, B. Zhang, X. Sun, Y. Yang, D. Lu, J. Yang, L.-L. Deng, L. Sun, H. Xu, S.-Y. Xie and Q. Zhang, Shell Dependence of Highly Tunable Circular Dichroism in Chiral Hybrid Plasmonic Nanomaterials for Chiroptical Applications, *ACS Nano*, 2025, **19**, 2961–2074.
- 78 F. Wang, W. Yang, Q. Ding, X. Xing, I. Xu, H. Lin, C. Xu and S. Li, Chiral Au@CeO₂ Helical Nanorods with Spatially Separated Structures for Polarization-dependent N₂ Photofixation, *Angew. Chem., Int. Ed.*, 2024, e202415031.
- 79 C. L. Hao, L. G. Xu, W. Ma, X. L. Wu, L. B. Wang, H. Kuang and C. L. Xu, Unusual Circularly Polarized Photocatalytic Activity in Nanogapped Gold-Silver Chiroplasmonic Nanostructures, *Adv. Funct. Mater.*, 2015, **25**, 5816–5822.
- 80 Y. Negrin-Montecelo, A. Movsesyan, J. Gao, S. Burger, Z. M. M. Wang, S. Nlate, E. Pouget, R. Oda, M. Comesana-Hermo, A. O. Govorov and M. A. Correa-Duarte, Chiral Generation of Hot Carriers for Polarization-Sensitive Plasmonic Photocatalysis, *J. Am. Chem. Soc.*, 2022, **144**, 1663–1671.
- 81 W. Fu, Q. Gao, C. Zhang, L. Tan, B. Jiang, C. Xiao, M. Liu and P.-p. Wang, Exploring Geometric Chirality in Nanocrystals for Boosting Solar-to-Hydrogen Conversion, *Angew. Chem., Int. Ed.*, 2024, **63**, e202411871.
- 82 G. Yang, L. Sun, Y. Tao, Q. Cheng, X. Sun, C. Liu and Q. Zhang, Chiral AuCu heterostructures with site-specific geometric control and tailored plasmonic chirality, *Sci. China Chem.*, 2023, **66**, 3280–3289.

