


 Cite this: *RSC Adv.*, 2025, 15, 42761

Investigation of electrical transport mechanisms in an n-CdIn₂Se₄/Pt thin film Schottky diode fabricated by pulsed laser deposition

 S. D. Dhruv,^a Tanvi Dudharejiya,^b Sergei A. Sharko,^c Aleksandra I. Serokurova,^c Nikolai N. Novitskii,^c D. L. Goroshko,^d Parth Rayani,^{ae} Jagruti Jangale,^a J. H. Markna,^b Bharat Kataria^b and D. K. Dhruv^{id}*^a

This investigation focuses on the fabrication of an n-CdIn₂Se₄/Pt thin film Schottky diode using the pulsed laser deposition technique. The typical grazing incidence X-ray diffractogram displays a sharp and bright (1 1 1) characteristic reflection, confirming the formation of polycrystalline CdIn₂Se₄ thin films. Various microstructural parameters have been calculated for the CdIn₂Se₄ thin films using the most prominent (1 1 1) reflection. Hall measurement examination confirmed the n-type conductivity of the CdIn₂Se₄ thin films. Characterization of the voltage–current curve of the vacuum-fabricated n-CdIn₂Se₄/Pt thin film Schottky diode confirms the presence of a typical Schottky diode-type junction between CdIn₂Se₄ and platinum with a good rectification ratio. The principal conducting mechanism of the produced n-CdIn₂Se₄/Pt thin film Schottky diode is thermionic emission at lower applied biases (≤ 0.5 V), while the space charge limited conduction mechanism is dominant at higher biases (> 0.5 V). The ideality factor values for the n-CdIn₂Se₄/Pt thin film Schottky diode are in the range of 1.4819 to 1.8102, depending on the temperature ($300 \text{ K} \leq T \leq 342 \text{ K}$). The zero-bias barrier height and effective Richardson's constant of the n-CdIn₂Se₄/Pt thin film Schottky diode are $\approx 0.8652 \text{ eV}$ and $\approx 1.8771 \times 10^5 \text{ A m}^{-2} \text{ K}^2$, respectively. The effective density of permitted energy levels is $\approx 1.5491 \times 10^{24} \text{ m}^{-3}$ in the conduction band of the n-CdIn₂Se₄ thin films. Additionally, characterization of the voltage–capacitance curve of the n-CdIn₂Se₄/Pt thin film Schottky diode revealed its zero bias built-in diffusion potential ($\approx 0.8178 \text{ V}$), donor impurity concentration ($\approx 5.9132 \times 10^{21} \text{ m}^{-3}$), and flat-band barrier height ($\approx 0.9525 \text{ eV}$). Based on Anderson's model, several electrical transport parameters were applied to depict the theoretical energy band diagram of the n-CdIn₂Se₄/Pt thin film Schottky diode. The functional groups present in the CdIn₂Se₄ thin films deposited on a platinum thin film substrate were determined using Fourier transform infrared spectroscopy.

 Received 5th August 2025
 Accepted 22nd October 2025

DOI: 10.1039/d5ra05715a

rsc.li/rsc-advances

1. Introduction

The first systematic study on metal–semiconductor rectifying systems is typically credited to Braun in 1874, who observed that the total resistance of a point contact depends on the specific surface characteristics and the polarity of the applied voltage.¹ Point contact rectifiers have been used practically in various

forms since the turn of the nineteenth century. Metal semiconductor contacts have been the subject of much research due to their significance in direct current and microwave applications and their intricate role in other semiconductor devices.² In particular, Schottky diode structures have been employed in solar cells as the gate electrode of metal-semiconductor field-effect transistors, photodetectors, and other applications. Above all, the metal contact on strongly doped semiconductors creates an ohmic contact, which is necessary to allow current to flow in and out of all semiconductor devices.³

The photoelectrical properties of an Au–CdIn₂S₄ surface barrier diode were studied by S. I. Radautsan *et al.*⁴ The current transport in p-type CdIn₂Te₄ Schottky diodes was studied by S. Kianian *et al.*⁵ The fabrication and electrical characterization of an Al/p-ZnIn₂Se₄ thin film Schottky diode structure were reported by Dhruv *et al.*⁶ Chong Ouyang *et al.* reported the use of a hierarchical MoO₂/ZnIn₂S₄ Schottky heterojunction to stimulate photocatalytic H₂ evolution under visible light.⁷

^aNatubhai V. Patel College of Pure and Applied Sciences, The Charutar Vidya Mandal (CVM) University, Vallabh Vidyanagar, 388120, Anand, Gujarat, India. E-mail: dhananjaydhruv@rediffmail.com

^bDepartment of Nanoscience and Advanced Materials, Saurashtra University, Rajkot 360005, Gujarat, India

^cLaboratory of Magnetic Films Physics, Scientific-Practical Materials Research Centre of National Academy of Sciences of Belarus, 220072 Minsk, Belarus

^dInstitute of Automation and Control Processes Far Eastern Branch of the Russian Academy of Sciences, 5 Radio St., Vladivostok 690041, Russia

^eGovernment Science College, Maharaja Krishnakumarsinhji Bhavnagar University, Gariyadhar, 364505, Bhavnagar, Gujarat, India



CdIn₂Se₄ thin films have been deposited using a range of techniques by different authors, including sol-gel dip coating, hydro evaporation, spray pyrolysis, chemical bath deposition, electrochemical, pulsed electrodeposition, and potentiostatic cathodic electrodeposition.^{8–14} Among them, pulsed laser deposition is a great option for depositing ternary semiconducting materials given that its parts have different vapor pressures. Compared to conventional vacuum-based thin film deposition techniques, pulsed laser deposition technology shortens the processing time by facilitating rapid thin film growth, while maintaining homogeneity. The most exciting aspect of pulsed laser deposition technology is its easy manufacture of multilayer devices and heterojunctions, and fine film thickness control. To create CdIn₂Se₄ thin films, scientists used pulsed laser deposition given that it has several advantages over traditional thin film deposition techniques, such as maintaining the precise composition of the target material in the formed film. Our extensive review and experimental work on CdIn₂Se₄ and/or CdIn₂Se₄ thin films^{15–17} shows that Hahn *et al.*¹⁸ initially reported work on this ternary semiconducting compound in 1955. After that, no one documented any work on CdIn₂Se₄ thin films until 1991. Our rigorous assessment indicates that no one has attempted to construct n-CdIn₂Se₄/Pt thin film Schottky diodes using pulsed laser deposition.

The present investigation focuses on optimizing the pulsed laser deposition parameters for the formation of single-phase, polycrystalline, stoichiometric CdIn₂Se₄ thin films. The crystal structure and microstructural parameters of CdIn₂Se₄ thin films deposited on amorphous quartz glass substrates previously pre-deposited with platinum substrates have been presented. In the case of the CdIn₂Se₄ thin film and/or n-CdIn₂Se₄/Pt thin film Schottky diode, characterization of their electrical transport was performed to determine their rectification ratio, reverse saturation current, ideality factor, zero-bias barrier height, effective Richardson's constant, effective density of allowed energy states in the conduction band, zero bias built-in diffusion potential, donor impurity concentration, flat-band barrier height, type of conductivity, *etc.* Furthermore, the purity of the CdIn₂Se₄ thin films was examined using Fourier transform infrared spectroscopy.

2. Experimental

2.1 Fabrication of n-CdIn₂Se₄/Pt thin film Schottky diode

A schematic illustration of the n-CdIn₂Se₄/Pt thin film Schottky diode device under investigation is shown in Fig. 1. A high-purity ($\approx 99.998\%$) platinum metal (CAS number: 7440-06-4; Merck) thin film with a thickness of about ≈ 100 nm was deposited at room temperature (≈ 300 K) on an ultrasonically cleaned amorphous quartz glass (fused silica) (Blue Star, Polar Industries Corporation, India) substrate employing an electron beam evaporation system (model: Auto 500; Hind High Vacuum Co. Pvt. Ltd, India). Using high-purity conductive silver adhesive paste (model: RS pro-RS186-3600; RS Components & Controls Limited, India), a single-phase CdIn₂Se₄ (ref. 16) pellet target and amorphous quartz glass substrates pre-deposited with platinum were mounted on the target holder and temperature-

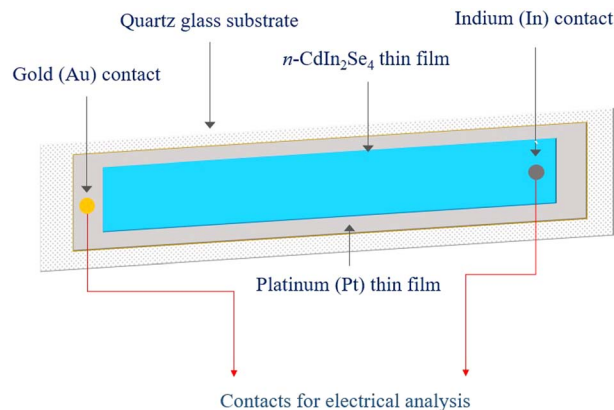


Fig. 1 The structure of the n-CdIn₂Se₄/Pt thin film Schottky diode (top view).

controlled substrate holder, respectively, in a stainless-steel pulsed laser deposition chamber (Excel Instruments, India). An ≈ 100 nm-thick CdIn₂Se₄ thin film was then deposited on the pre-deposited platinum thin film substrate at a substrate temperature of ≈ 550 K using a high-vacuum pulsed laser deposition (model: Compex-Pro Excimer Laser 102F; Coherent, Germany) technique by irradiating the CdIn₂Se₄ pellet target with a krypton fluoride pulsed laser. The details of the thin film deposition technique are elaborated elsewhere.¹⁵ Using a suitable mica mask arrangement, the effective cross-sectional area of the n-CdIn₂Se₄/Pt thin film Schottky diode device was set to $\approx 2.5000 \times 10^{-5} \text{ m}^2$ (≈ 8.3 mm length \times 3.0 mm breadth).

Our previous investigation¹⁵ on the deposition and transport characterization of CdIn₂Se₄ thin films suggests that their DC electrical resistivity decreases with an increase in the substrate temperature. The DC electrical resistivity reached the lowest value of $\approx 0.1751 \Omega \text{ m}$ at a substrate temperature of ≈ 550 K. Therefore, in the current study, thin films of CdIn₂Se₄ were deposited on amorphous quartz glass substrates pre-deposited with platinum at the optimal substrate temperature of ≈ 550 K.

To produce stoichiometric polycrystalline CdIn₂Se₄ thin films, several pulsed laser deposition system parameters have been customized, as indicated in Table 1.

2.2 Characterization of n-CdIn₂Se₄/Pt thin film Schottky diode

The crystal structure of the CdIn₂Se₄ thin films deposited on amorphous quartz glass substrates pre-deposited with platinum

Table 1 Optimized pulsed laser deposition parameters for the synthesis of the CdIn₂Se₄ thin films

Sr. no.	Parameter	Value
01	KrF laser wavelength (λ)	≈ 248 nm
02	KrF laser energy	≈ 249 mJ
03	Repetition rate (frequency)	≈ 4 Hz
04	Source-to-substrate distance	≈ 45 mm
05	Base pressure (vacuum)	$\approx 1.5000 \times 10^{-4}$ Pa
06	Deposition time (laser ablation time)	≈ 4 min
07	Evaporation rate	$\approx 10 \text{ nm s}^{-1}$



using the pulsed laser deposition technique at a substrate temperature of ≈ 550 K was examined using a grazing incidence X-ray diffractometer (model: SmartLab 9 kW; Rigaku Corporation, Japan) with the typical $\text{CuK}\alpha$ radiation at a wavelength of ≈ 0.1541 nm in the 2θ diffraction angle range of 20° to 80° . The electrical transport characterization of the n- CdIn_2Se_4 /Pt thin film Schottky diode was performed using a DC power supply (model: IT6722A; ITech, Taiwan), a digital multimeter (model: DMM 6500 6.5; Keithley, USA), and a semiconductor characterization system (model: SCS-4200; Keithley, USA). At room temperature (≈ 303 K), a Fourier transform infrared spectrophotometer (model: IRSpirit-X; Shimadzu, Japan) operated in mid-infrared mode in the wavenumber range of $4000\text{--}600$ cm^{-1} with a resolution of ≈ 2 cm^{-1} was employed to identify the presence of the functional groups in the CdIn_2Se_4 thin films deposited on amorphous quartz glass substrates that had been pre-deposited with platinum at a substrate temperature of ≈ 550 K.

The electrical transport characterization of the n- CdIn_2Se_4 /Pt thin film Schottky diode as a function of temperature ($300\text{ K} \leq T \leq 342\text{ K}$) was conducted inside a glass dome at a low vacuum of ≈ 1.40 Pa, created using a rotary pump (model: FD12; Hind High Vacuum Co. Pvt. Ltd, India) to prevent contamination from the ambient conditions. All the pulsed laser deposition parameters listed in Table 1 were kept almost constant during

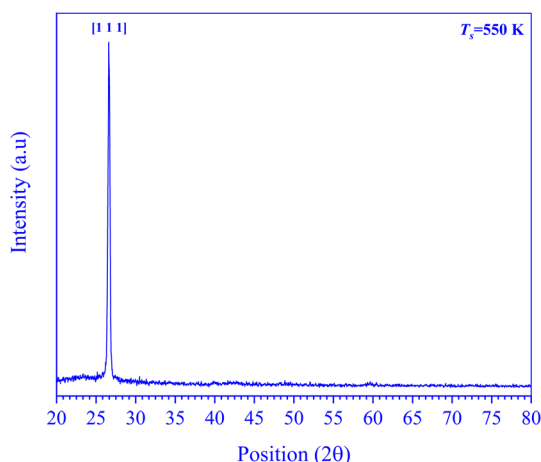


Fig. 2 Typical grazing incidence X-ray diffractogram of the CdIn_2Se_4 thin film.

the fabrication of the n- CdIn_2Se_4 /Pt thin film Schottky diode to guarantee the reproducibility of the results, as several pre- and post-deposition parameters can influence the properties of thin films in vacuum technology. The present study analyzed, interpreted, and reported the statistically significant data from the simultaneous fabrication of several n- CdIn_2Se_4 /Pt thin film Schottky diodes under identical conditions.

3. Results and discussion

3.1 Grazing incidence X-ray diffraction (GI-XRD) analysis

The typical grazing incidence X-ray diffractogram in Fig. 2 shows a sharp and intense $(1\ 1\ 1)$ characteristic reflection at a diffraction angle of $\approx 26.60^\circ$, indicating that the CdIn_2Se_4 thin films deposited on amorphous quartz glass substrates pre-deposited with platinum at a substrate temperature of ≈ 550 K are polycrystalline; the results are in good agreement with the ICDD 01-089-2388 database for tetragonal (pseudo-cubic) α -phase CdIn_2Se_4 with space group $P42m$ $(1\ 1\ 1)$ reflections, showing no signs of an elemental or secondary phase.^{15,16}

Table 2 displays the microstructural parameters including lattice constant, cell volume, d -interplanar spacing, stacking fault, crystallite size, dislocation density, and lattice strain calculated from the grazing incidence X-ray diffractogram pattern for the most noticeable $(1\ 1\ 1)$ reflection of the α - CdIn_2Se_4 thin films on amorphous quartz glass substrates that had previously been pre-deposited with platinum using the pulsed laser deposition technique at a substrate temperature of ≈ 550 K.¹⁹

Using eqn (1) for a tetragonal crystal structure, the lattice constant (a) of the α - CdIn_2Se_4 thin film (where $a \approx c$) was determined,²⁰ as follows:

$$\frac{1}{a^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2} \quad (1)$$

In the tetragonal crystal structure, the unit cell volume (V) of the α - CdIn_2Se_4 thin film can be inferred using the relation $V = a^2c$.²¹

Using Bragg's eqn (2), the d -interplanar spacing value of the α - CdIn_2Se_4 thin film was determined,²² as follows:

$$n\lambda = 2d \sin \theta \quad (2)$$

Planar flaws known as stacking faults (SF) disturb the normal arrangement of atomic layers. A higher stacking fault

Table 2 Microstructural parameters of the α - CdIn_2Se_4 thin film

Microstructural parameters	Reported value	Current investigation	Deviation (%)	Reference/Method
Lattice constant (a) (nm)	0.5815	0.5801	0.2408	ICDD card 01-089-2388
Cell volume (nm^3)	0.1966	0.1952	0.7121	ICDD card 01-089-2388
d_{111} (nm)	0.3357	0.3349	0.2383	ICDD card 01-089-2388
Stacking fault ($\times 10^{-3}$)	—	4.62	—	—
Crystallite size (nm)	—	16.08	—	Scherrer method
Dislocation density ($\times 10^{-3}$) (lines per nm^2)	—	3.87	—	Williamson–Smallman's method
Lattice strain ($\times 10^{-3}$)	—	9.37	—	—



value introduces localized energy states inside the bandgap, changing the leakage behavior and current flow. Furthermore, a greater stacking fault value may result in a decrease in the rectification efficiency and an increase in the reverse leakage current. The stacking fault value of the α -CdIn₂Se₄ thin film can be obtained using eqn (3).²³

$$SF = \frac{0.2533\beta}{(\tan \theta)^2} \quad (3)$$

where β represents the full breadth/width at half maximum (FWHM) of the diffraction peak in the above-mentioned relation, enabling the determination of the crystallite size (D) of the tetragonal α -phase CdIn₂Se₄ thin film using the Scherrer eqn (4),²⁴ as follows:

$$D\beta \cos \theta = K\lambda \quad (4)$$

Dislocation density (δ) is a defect that arises from misaligned atoms in the crystal lattice. It shortens the carrier lifespan by acting as a location for electron and hole recombination. It increases the leakage current, while lowering the breakdown voltage. It could lead to less-than-ideal diode behavior by making the Schottky barrier less uniform. A high dislocation density leads to low device reliability and thermal instability. The dislocation density of the α -CdIn₂Se₄ thin film can be obtained by substituting the crystallite size value obtained from the Scherrer method into eqn (5).

$$\delta = \frac{1}{D^2} \quad (5)$$

Strain (ε) is caused by lattice mismatches or differences in thermal expansion between the substrate and the thin film. Strain alters the band structure and carrier mobility. Consequently, uncontrolled strain typically degrades the transport properties. The Stokes–Wilson (S–W) eqn (6) can be employed to determine the lattice strain developed in the α -CdIn₂Se₄ thin film due to crystal imperfection and disorder.²⁵

$$\beta = 4\varepsilon \tan \theta \quad (6)$$

The low dislocation density, stacking fault, and lattice strain in the fabricated n-CdIn₂Se₄/Pt thin film Schottky diode could be the factors responsible for the enhanced performance of the thin film Schottky diode device in this work.

3.2 Hall effect analysis

The CdIn₂Se₄ thin film produced at a substrate temperature of ≈ 550 K consistently exhibited n-type conductivity, according to hot probe tests. When recorded at a magnetic field of ≈ 0.2001 T, Hall measurements yielded a negative Hall coefficient value of $\approx -0.4488 \times 10^{-3} \text{ m}^3 \text{ C}^{-1}$, which is consistent with the findings of hot probe testing and suggests that the majority of charge carriers are electrons.²⁶ According to the Hall measurements, the donor impurity concentration in the n-type CdIn₂Se₄ thin films is $\approx 7.1300 \times 10^{21} \text{ m}^{-3}$.¹⁵

3.3 Voltage–current characteristics

Fig. 3 shows the sweep forward and reverse biased electrical transport characterization of the n-CdIn₂Se₄/Pt thin film Schottky diode at various ambient temperatures ($300 \text{ K} \leq T \leq 342 \text{ K}$). The rectifying electrical transport characteristic evident in Fig. 3 confirms the formation of a typical diode-type junction between n-CdIn₂Se₄/Pt, signifying the formation and presence of a depletion width between the n-type CdIn₂Se₄ semiconducting material and tungsten metal, which in turn restricts the flow of forward- and reverse-charge carriers at the intersection region.²⁷

According to Table 3, the rectification ratio of the n-CdIn₂Se₄/Pt thin film Schottky diode increases when the ambient temperature decreases ($300 \text{ K} \leq T \leq 342 \text{ K}$) and the bias is constant; when the ambient temperature remains constant, the rectification ratio increases as the bias increases. Constrained deficiencies in the intersection zone and the increase in the leakage current with an increase in ambient temperature may be two ways to describe this effect.^{28,29}

The effective evaluation of the current conduction processes in all manufactured semiconducting thin film electronic devices requires a detailed examination of their electrical transport characteristics. The electrical transport characterization reveals two current conduction regions, *i.e.*, a rectifying region at a higher applied bias and a practically ohmic region at a lower applied bias. Ohmic behaviour results from the inserted charge carrier density being less than the thermally generated carrier density at a lower applied bias. The trap-controlled space charge limiting current/conduction mechanism predominates in this region given that the slope is greater than two with a higher applied bias.²⁸

To learn more about the current conduction mechanism in the n-CdIn₂Se₄/Pt thin film Schottky diode at a lower applied bias (≤ 0.5 V), as well as to determine the reverse saturation current and ideality factor, the model put forth by Sze and

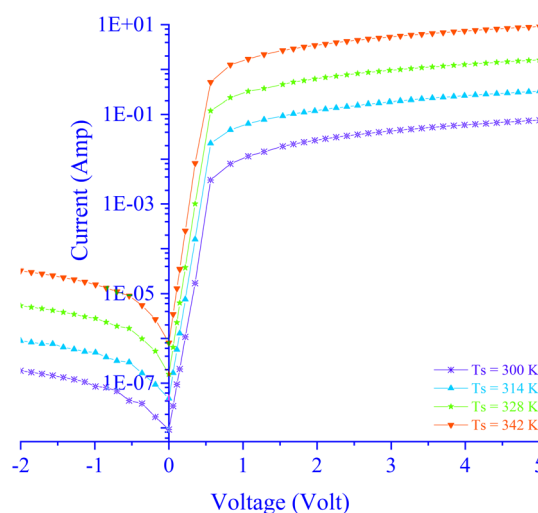


Fig. 3 Electrical transport characterization of the n-CdIn₂Se₄/Pt thin film Schottky diode.



Table 3 Rectification ratio of the n-CdIn₂Se₄/Pt thin film Schottky diode

Temperature (K)	Rectification ratio ($\times 10^5$)							
	Applied bias (V)		Applied bias (V)		Applied bias (V)		Applied bias (V)	
	-2.0011	+1.9690	-1.5682	+1.5362	-0.9904	+1.0705	-0.5371	+0.5638
300	1.4178		1.3328		1.2762		0.7974	
314	1.3770		1.2525		1.1936		0.7381	
328	1.1584		1.1152		1.0689		0.6901	
342	1.0798		1.0672		0.9914		0.5489	

Crowell³ was used and fitted in the current investigation. Eqn (7) illustrates how the applied bias alters the current.

$$\frac{I}{I_s} = \exp\left(\frac{qV}{nkT}\right) \quad (7)$$

In eqn (7), $\frac{kT}{q}$ is the thermal voltage (≈ 0.0259 V) at ≈ 300 K.

Fig. 4 shows the variation in natural logarithmic current for a constructed n-CdIn₂Se₄/Pt thin film Schottky diode at different ambient temperatures with an applied bias (≤ 0.5 V). With an increase in the applied bias (≤ 0.5 V), the forward current of the constructed n-CdIn₂Se₄/Pt thin film Schottky diode increases exponentially, as demonstrated by the linear-straight-line curve in Fig. 4. Also, the linear-straight-line plot indicates that the main characteristic of the produced n-CdIn₂Se₄/Pt thin film Schottky diode is thermionic emission at a lower applied voltage (≤ 0.5 V).⁶ The ideality factors of the n-CdIn₂Se₄/Pt thin film Schottky diode at different ambient temperatures were determined using the slope values, and the reverse saturation currents were derived from the linear-straight-line intercept values shown in Fig. 4.

The reverse saturation current and ideality factor at various ambient temperatures are displayed in Table 4.

In a thin film Schottky diode made of metal and semiconductor, its reverse saturation current mainly depends on the ambient temperature $\left[I_s = T^3 \exp\left(-\frac{E_g}{kT}\right) \right]$, whereas it is not affected by the reverse voltage.³

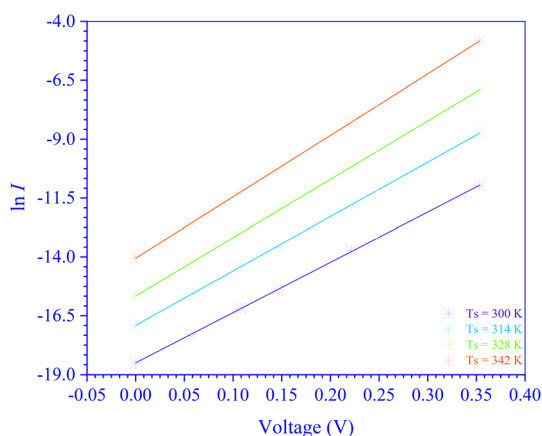


Fig. 4 The variation in $\ln I$ with applied bias for the n-CdIn₂Se₄/Pt thin film Schottky diode.

Temperature has a much more noticeable effect because thin films have greater grain boundaries and defect concentrations, which encourage the generation of carriers.³⁰ Consequently, the reverse saturation current decreases as the ambient temperature decreases because fewer minority charge carriers are in the depletion width. The ideality factor of a diode shows how well it follows the ideal diode formula. The ideality factor value typically ranges from one, which indicates ideal diffusion, to two, which indicates dominant recombination. Because recombination predominates in the depletion region, the ideality factor values get closer to two at lower temperatures. As the temperature increases, the ideality factor gets closer to one because recombination decreases and the diffusion current increases. The ideality factor value is observed at low and/or high ambient concentrations; however, it substantially changes as the doping concentration increases and/or the ambient temperature decreases. The ideality factor shifts in thin films at high temperatures due to the increased carrier mobility and decreased trap-assisted recombination.³¹

Eqn (8) was utilized to analyze the current conduction mechanism in the n-CdIn₂Se₄/Pt thin film Schottky diode at a lower applied bias (≤ 0.5 V), as well as to determine the zero-bias barrier height and effective Richardson's constant values.³²

$$\frac{AA^*T^2}{I_s} = \exp\left(\frac{q\phi_b}{kT}\right) \quad (8)$$

In eqn (8), A is the effective cross-sectional area of the n-CdIn₂Se₄/Pt thin film Schottky diode. The effective Richardson's constant $\left(A^* = \frac{4\pi qm_n k^2}{h^3} \right)$ is an effective mass-dependent parameter.

The linear-straight-line character of Fig. 5 suggests a thermionic emission conduction mechanism in the n-CdIn₂Se₄/Pt thin film Schottky diode at lower applied voltages (< 0.5 V).

Table 4 Temperature-dependent electrical parameters of the n-CdIn₂Se₄/Pt thin film Schottky diode

Temperature (K)	Intercept	Reverse saturation current ($\times 10^{-9}$) (A)	Slope	Ideality factor
300	-18.50560	9.1859	21.3617	1.8102
314	-16.90890	45.3480	23.1085	1.6734
328	-15.66250	157.7100	24.7977	1.5594
342	-14.05760	784.9900	26.0940	1.4819



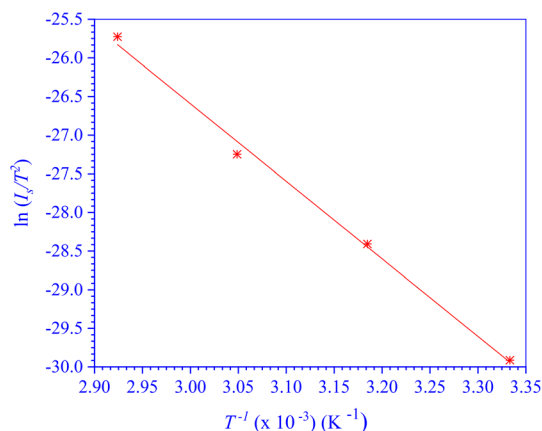


Fig. 5 $\ln\left(\frac{I_s}{T^2}\right)$ versus $\frac{1000}{T}$ plot of the n-CdIn₂Se₄/Pt thin film Schottky diode.

The zero-bias barrier height of the n-CdIn₂Se₄/Pt thin film Schottky diode is determined using the slope in Fig. 5 and found to be ≈ 0.8652 eV. The intercept extracted from Fig. 5 gives the effective Richardson's constant $\approx 1.8771 \times 10^5$ A m⁻² K² for the CdIn₂Se₄ thin films grown at a substrate temperature of ≈ 550 K, which is equivalent to $0.1562 m_0$. The room temperature effective mass values of n-type CdIn₂Se₄ reported by various other authors are $0.15 m_0$,^{33,34} $0.1600 m_0$,³⁴ and $0.1780 m_0$.³⁵

By inserting the effective mass of n-type CdIn₂Se₄ and other commonly reported values, it is possible to determine that the effective density of allowed energy states in the conduction

band $\left(= 2 \left[\frac{2\pi m_n kT}{h^2} \right]^{\frac{3}{2}} \right)$ is similarly an effective mass-

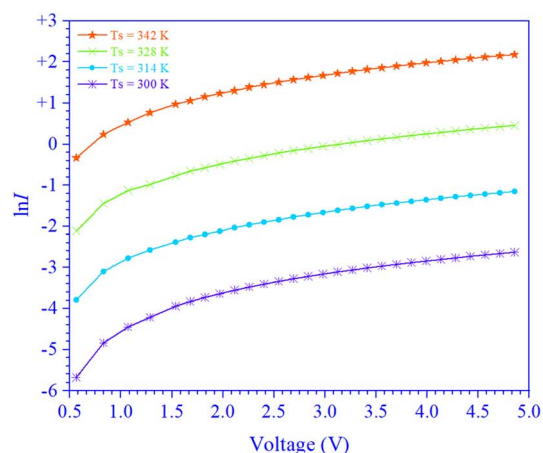
dependent characteristic, which is $\approx 1.5491 \times 10^{24}$ m⁻³. The deduced effective density of allowed energy states in the conduction band is consistent with the values published by Bhalerao *et al.*¹⁰

To verify the current conduction mechanism at a higher bias (>0.5 V), a natural logarithmic current was recorded against applied bias (>0.5 V) at varying ambient temperatures, as shown in Fig. 6(a). At a greater bias (>0.5 V), the rectifying nature of the plot in Fig. 6(a) suggests a variable current conduction mechanism for the n-CdIn₂Se₄/Pt thin film Schottky diodes.

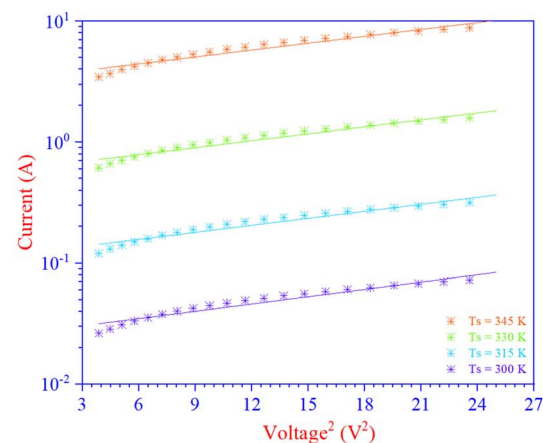
Fig. 6(b) illustrates the change in current against the square of the applied bias at a greater applied bias (>0.5 V). Fig. 6(b) shows straight line character, supporting the space charge restricted current/conduction mechanism of the n-CdIn₂Se₄/Pt thin film Schottky diode.²⁸ The space charge restricted current/conduction mechanism is dominated by charge carriers injected from the contacts; the current is solely dependent on mobility and not on the density of the charge carriers, and the voltage-current characteristics follow a second-degree polynomial relation.

3.4 Voltage-capacitance characteristics

The change in capacitance with applied bias was recorded at a frequency of 500 kHz at room temperature to extract several



a)



b)

Fig. 6 The variation in current with applied bias (>0.5 V) for the n-CdIn₂Se₄/Pt thin film Schottky diode.

important parameters, including the zero-bias built-in diffusion potential, donor impurity concentration, flat-band barrier height, and type of conductivity of the n-CdIn₂Se₄ thin film and/or n-CdIn₂Se₄/Pt thin film Schottky diode. Eqn (9) establishes an empirical relationship between applied bias and junction capacitance.³²

$$\frac{1}{2C^2} = \frac{(V - V_0 + \frac{kT}{q})}{q\epsilon_0\epsilon_r N_D A^2} \quad (9)$$

In eqn (9), V_0 is the zero bias built-in diffusion potential (≈ 0.8178 V), which may be possibly due to the non-homogeneous surface states in the system, ϵ_r is the relative dielectric constant of n-CdIn₂Se₄ (≈ 6.2000 (ref. 36)), and N_D is the donor impurity concentration.

The donor impurity concentration of the n-CdIn₂Se₄ thin films is $\approx 5.9132 \times 10^{21}$ m⁻³, as inferred from the slope $\left(\approx 6.1529 \times 10^{16} \frac{F^{-2}}{V} = \frac{2}{q\epsilon_0\epsilon_r N_D A^2} \right)$ of the inverse of the square of capacitance vs. applied bias plot in Fig. 7. The donor impurity



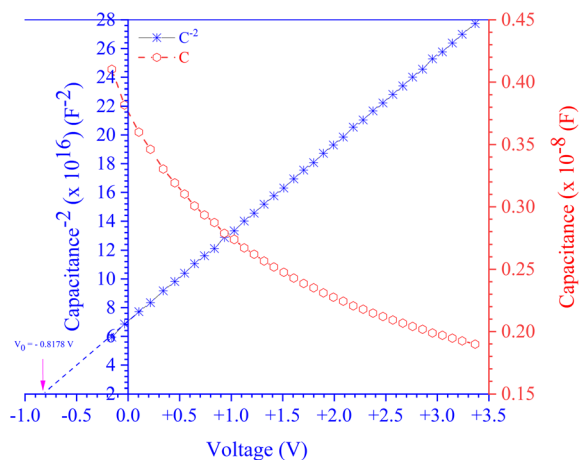


Fig. 7 Variation in capacitance as a function of the bipolar voltage for the n-CdIn₂Se₄/Pt thin film Schottky diode.

concentration ($\approx 7.1300 \times 10^{21} \text{ m}^{-3}$) derived from the voltage–capacitance measurements and Hall effect tests is in close agreement with the donor impurity concentration in n-CdIn₂Se₄ thin films reported by Bhalerao *et al.*¹⁰

One can use eqn (10) to determine the flat-band barrier height of the n-CdIn₂Se₄/Pt thin film Schottky diode,²⁸ as follows:

$$\left(\phi_b - V_0 - V_n = \frac{kT}{q} - \Delta\phi_B \right) \quad (10)$$

In eqn (10), V_n ($\approx E_C - E_{F_n}$) is the potential difference between the conduction band (CB) (E_C) and Fermi level (E_{F_n}) in n-type CdIn₂Se₄. Using eqn (11),³⁷ the value of V_n is calculated to be $\approx 0.1440 \text{ V}$, which is extremely close to the V_n ($\approx 0.1430 \text{ V}$) value reported by Bhalerao *et al.*¹⁰

$$\frac{N_C}{N_D} = \exp\left(\frac{qV_n}{kT}\right) \quad (11)$$

It is necessary to consider an appropriate interface electric field when a metal and a semiconductor come into contact, and to substitute the dielectric constant of the semiconductor (≈ 6.2000 for n-CdIn₂Se₄)³⁶ for the free space dielectric constant. Eqn (12) expresses the reduction in the Schottky barrier height.⁶

When a metal is in contact with a semiconductor, the appropriate interface electric field has to be considered, and the free space dielectric constant has to be replaced by the dielectric constant of the semiconductor (≈ 6.2000 (ref. 36) for n-CdIn₂Se₄). The Schottky barrier image force barrier lowering can be expressed by employing eqn (12),³

$$\Delta\phi_B = \left(\frac{qE_m}{4\pi\epsilon_0\epsilon_s} \right)^{\frac{1}{2}} \quad (12)$$

Eqn (13) can be used to deduce the maximum electric field at the n-CdIn₂Se₄/Pt thin film Schottky diode junction. The derived value is $\approx 5.3129 \times 10^6 \text{ V m}^{-1}$.³

$$E_m = \left(\frac{2qN_D V_0}{\epsilon_0\epsilon_s} \right)^{\frac{1}{2}} \quad (13)$$

In the case of the n-CdIn₂Se₄/Pt thin film Schottky diode, its Schottky barrier image force barrier lowering value deduced from eqn (12) is $\approx 3.5128 \times 10^{-2} \text{ V}$.

By substituting $V_n \approx 0.1440 \text{ V}$, $V_0 \approx 0.8178 \text{ V}$, and $\Delta\phi_B$ into eqn (10), the flat-band barrier height for the n-CdIn₂Se₄/Pt thin film Schottky diode is found to be $\approx 0.9525 \text{ eV}$.

In contrast to the flat-band barrier height result from the voltage–capacitance experiment ($\approx 0.9525 \text{ eV}$), the zero-bias barrier height from the voltage–current measurement ($\approx 0.8652 \text{ eV}$) is tiny. This discrepancy in the barrier height results between the two approaches may be due to the edge leakage current, deep impurity levels, or interface contamination.³⁸

3.5 Energy band diagram construction for n-CdIn₂Se₄/Pt thin film Schottky diode

The conductance of current and its capacitance behaviour are controlled by the barrier at the metal–semiconductor interface when metal comes into contact with a semiconductor.³ The fundamental energy band diagram that forms the barrier height and certain effects that can alter its value are examined in this section.

Firstly, the ideal scenario is considered, which is free of anomalies such as interface states. The electrical energy relation between the n-type CdIn₂Se₄ semiconductor and high work function platinum metal, which are in different systems and not in contact, are depicted in Fig. 8(a).

Charge will move from the n-type CdIn₂Se₄ semiconducting material to the platinum metal when communication between them is permitted, as seen in Fig. 8(b), and thermal equilibrium is reached as a single system. Both the n-type CdIn₂Se₄ and platinum sides will have matching Fermi values. The contact potential, or difference between the two work functions, lowers the Fermi level in the n-type CdIn₂Se₄ semiconductor by a value equivalent to the Fermi level in the platinum metal. The work functions of the n-type CdIn₂Se₄ semiconducting material and platinum metal are $\approx 4.1858 \text{ eV}$ and $\approx 5.1200 \text{ eV}$, respectively.³³ A contact potential of $\approx 0.9342 \text{ eV}$ is found. The barrier height in the n-CdIn₂Se₄/Pt thin film Schottky diode is determined to be $\approx 1.0782 \text{ eV}$, which is determined by the difference between the platinum metal work function and the electron affinity of the n-type CdIn₂Se₄ semiconducting material ($\approx 4.0418 \text{ eV}$).

3.6 Fourier transform infrared (FTIR) spectroscopy analysis

In the present investigation, only 5N (99.999%) pure cadmium, indium, and selenium were used to synthesize CdIn₂Se₄.¹⁶ After that, CdIn₂Se₄ was deposited on amorphous quartz glass substrates that had already been coated with platinum using pulsed laser deposition, a non-chemical thin film deposition method.

Fig. 9 displays the Fourier transform infrared spectrum of the n-type CdIn₂Se₄ thin films deposited at a substrate temperature of $\approx 550 \text{ K}$ on amorphous quartz glass substrates



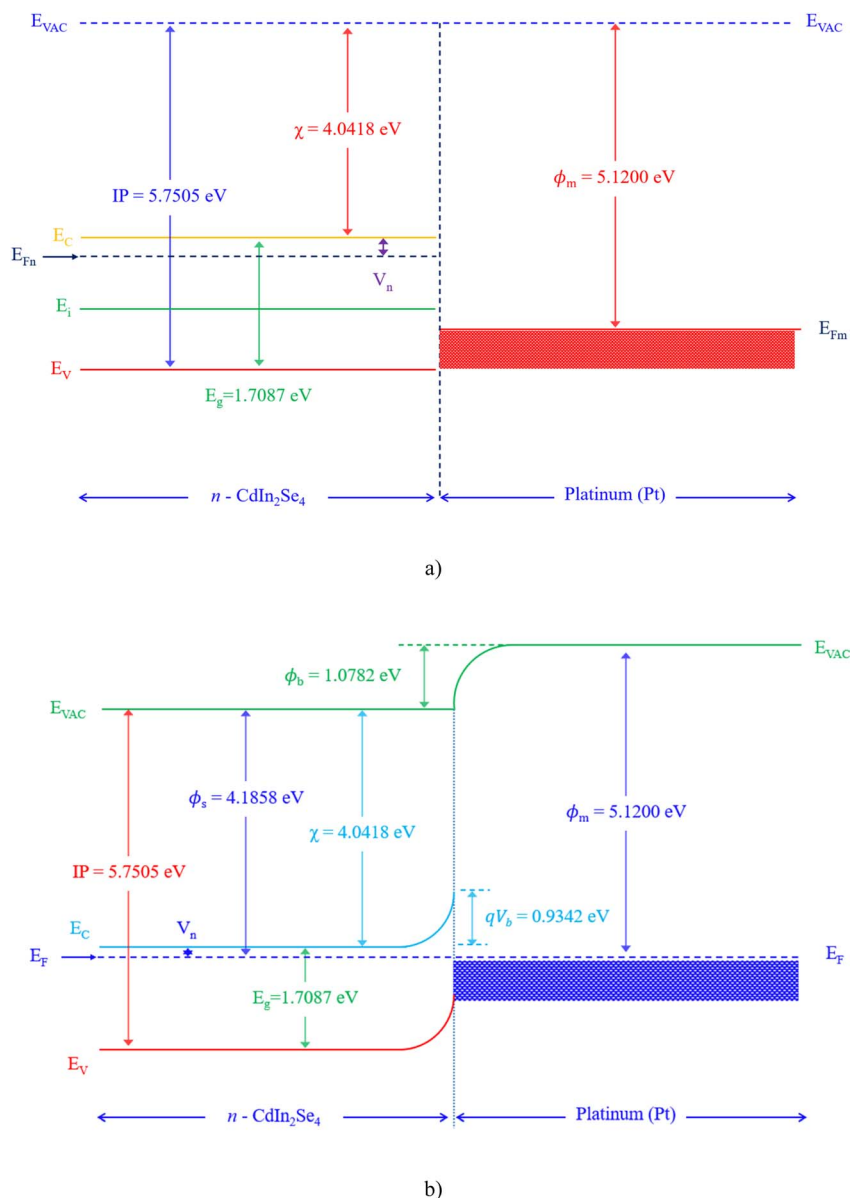


Fig. 8 Energy band diagram of the n-CdIn₂Se₄/Pt thin film Schottky diode: (a) isolated and (b) constructed.

that had been pre-deposited with platinum. The spectrum was recorded in the wavenumber ($\bar{\nu}$) range of 4000 to 400 cm^{-1} at room temperature (≈ 300 K). Cartesian coordinates have been used to represent the position of the peaks; for instance, 711.8757 indicates the wavenumber ($\bar{\nu}$) and 0.0148 denotes the absorbance.

The peak depicted at a wavenumber of ≈ 711.8757 cm^{-1} can be attributed to the alkene class bending C=C functional group. The fluoro compound class stretching C-F functional group is responsible for the peaks observed at wavenumbers of ≈ 1036.2383 cm^{-1} , ≈ 1266.5933 cm^{-1} , and ≈ 1387.8705 cm^{-1} . The nitro compound class stretching N-O functional group is responsible for the peak depicted at a wavenumber of ≈ 1516.3238 cm^{-1} . The aromatic compound class bending C-H functional group corresponds to the peaks detected at

wavenumbers of ≈ 1746.6788 cm^{-1} , ≈ 1905.9896 cm^{-1} , and ≈ 1990.6684 cm^{-1} . The isothiocyanate class stretching N=C=S functional group is responsible for the peaks detected at wavenumbers of ≈ 2076.0648 cm^{-1} and ≈ 2108.3575 cm^{-1} . The carboxylic class stretching O-H functional group can cause the peak detected at a wavenumber of ≈ 2524.5751 cm^{-1} . The stretching O-H functional group in the phenol and/or alcohol class explains the peaks observed at wavenumbers of ≈ 3532.8264 cm^{-1} , ≈ 3717.2539 cm^{-1} , and ≈ 3784.7099 cm^{-1} . The absorption peaks resulting from bending C=C, stretching C-F, stretching N-O, bending C-H, stretching N=C=S, and stretching O-H functional groups usually show strong and/or sharp absorption edges. It can be concluded that the CdIn₂Se₄ thin films deposited by pulsed laser deposition on amorphous quartz glass substrates pre-deposited with platinum at



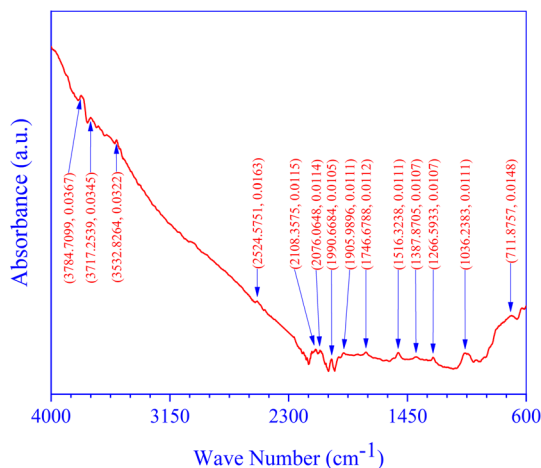


Fig. 9 Fourier transform infrared spectrum of the n-CdIn₂Se₄ thin film.

a substrate temperature of ≈ 550 K are pure and free of any functional groups because the absorption edges visible in the Fourier transform infrared spectrum of CdIn₂Se₄ are very weak.^{8,39}

4. Conclusions

Using a high-vacuum pulsed laser deposition approach, an ≈ 100 nm-thick CdIn₂Se₄ thin film was deposited on amorphous quartz glass substrates that had previously been pre-deposited with platinum at a substrate temperature of ≈ 550 K. A sharp and intense $\langle 111 \rangle$ characteristic reflection is seen in the typical grazing incidence X-ray diffractogram at a diffraction angle of $\approx 26.60^\circ$, suggesting the polycrystalline nature of the CdIn₂Se₄ thin films. The unit cell (≈ 0.5801 nm), cell volume (≈ 0.1952 nm³), d_{111} (≈ 0.3349 nm), stacking fault ($\approx 4.62 \times 10^{-3}$), crystallite size (≈ 16.08 nm), dislocation density ($\approx 3.87 \times 10^{-3}$ lines per nm²), and lattice strain ($\approx 9.37 \times 10^{-3}$) were all calculated for the CdIn₂Se₄ thin films from the most prominent $\langle 111 \rangle$ reflection in the GI-XRD. The n-type conductivity of the CdIn₂Se₄ thin films was verified by Hall measurement analysis. A typical Schottky diode-type junction between CdIn₂Se₄ and platinum with a fair rectification ratio is confirmed by the voltage–current characterization of the vacuum-fabricated n-CdIn₂Se₄/Pt thin film Schottky diode. At a lower applied bias (≤ 0.5 V), the primary conducting mechanism in the manufactured n-CdIn₂Se₄/Pt thin film Schottky diode is thermionic emission, whereas at a higher bias (> 0.5 V), the space charge limited conduction mechanism is dominant. The ideality factor values for the n-CdIn₂Se₄/Pt thin film Schottky diode vary from 1.4819 to 1.8102, depending on temperature ($300 \text{ K} \leq T \leq 342 \text{ K}$). The zero-bias barrier height and effective Richardson's constant were determined to be ≈ 0.8652 eV and $\approx 1.8771 \times 10^5 \text{ A m}^{-2} \text{ K}^2$ ($\approx 0.1562 m_0$), respectively, for the n-CdIn₂Se₄/Pt thin film Schottky diode. In the conduction band, the effective density of allowed energy states for the n-CdIn₂Se₄ thin films is $\approx 1.5491 \times 10^{24} \text{ m}^{-3}$. The voltage–capacitance characterization

of the n-CdIn₂Se₄/Pt thin film Schottky diode yielded its flat-band barrier height (≈ 0.9525 eV), donor impurity concentration ($\approx 5.9132 \times 10^{21} \text{ m}^{-3}$), and zero bias built-in diffusion potential (≈ -0.8178 V). Based on Anderson's model, different electrical transport parameters were employed to plot the theoretical energy band diagram for the n-CdIn₂Se₄/Pt thin film Schottky diode. The functional groups (if any) present in the CdIn₂Se₄ thin films deposited on a platinum thin film substrate were extracted using Fourier transform infrared spectroscopy.

Author contributions

All writers contributed to the conceptualization and design of the study. Material preparation, data gathering, and analysis were performed by S. D. Dhruv, Tanvi Dudharejiya, Sergei A. Sharko, Aleksandra I. Serokurova, Nikolai N. Novitskii, D. L. Goroshko, Parth Rayani, Jagruti Jangale, J. H. Markna, Bharat Kataria, and D. K. Dhruv. D. K. Dhruv wrote the first draft of the manuscript, and all authors commented on the previous versions. All authors read and approved the final manuscript. S. D. Dhruv: methodology. Tanvi Dudharejiya: data curation. Sergei A. Sharko: investigation. Aleksandra I. Serokurova: formal analysis. Nikolai N. Novitskii: visualization. D. L. Goroshko: writing – review & editing. Parth Rayani: resources. Jagruti Jangale: validation. J. H. Markna: software. Bharat Kataria: conceptualization. D. K. Dhruv: writing – original draft, supervision, project administration.

Conflicts of interest

The authors have no related financial or non-financial interests to disclose. The submitted work is the authors' original research and has not been communicated elsewhere for publication. On behalf of all the co-authors, I declare no conflict of interest.

Data availability

The authors declare that the data supporting the findings of this study are available/included in the main text of the paper.

Acknowledgements

This work was supported by the Department of Science & Technology (DST), Government of India, New Delhi (File number: DST/INT/BLR/P-36/2023) and the Belarusian Republican Foundation for Fundamental Research (File number: F23INDG-005) under the India-Belarus Programme of Co-operation in Science & Technology (Joint Research Project). The authors thank the Sophisticated Instrumentation Centre for Applied Research and Testing (SICART), Vallabh Vidyanagar-388120, Anand, Gujarat, India, for providing the FTIR spectrophotometer facility at a nominal rate.

References

- 1 F. Braun, in *Semiconductor Devices: Pioneering Papers*, World Scientific, 1991, pp. 377–380.



- 2 E. H. Rhoderick and R. H. Williams, *Metal-semiconductor Contacts*, Clarendon Press ; Oxford University Press, Oxford [England]: New York, 2nd edn, 1988.
- 3 S. M. Sze and K. K. Ng, *Physics of Semiconductor Devices*, Wiley, 1st edn, 2006.
- 4 S. I. Radautsan, I. P. Molodyan, S. V. Bulyarskii, N. N. Syrbu and V. E. Tezlevan, *Phys. Status Solidi A*, 1974, **21**, 617–622.
- 5 S. Kianian, S. A. Eshraghi, O. M. Stafsudd and A. L. Gentile, *J. Appl. Phys.*, 1987, **62**, 1500–1502.
- 6 D. K. Dhruv and B. H. Patel, *Mater. Sci. Semicond. Process.*, 2016, **54**, 29–35.
- 7 C. Ouyang, W. Huang, H. Tang, W. Liu, X. Gu, Z. Hong and M. Zhi, *ACS Appl. Energy Mater.*, 2022, **5**, 12739–12751.
- 8 A.-A. Ruanthon, T. Sarakonsri and C. Thanachayanont, *Funct. Mater. Lett.*, 2009, **02**, 199–203.
- 9 D. Sudha, S. Dhanapandian, C. Manoharan and A. Arunachalam, *Results Phys.*, 2016, **6**, 599–605.
- 10 A. B. Bhalerao, B. G. Wagh, R. N. Bulakhe, P. R. Deshmukh, J.-J. Shim and C. D. Lokhande, *J. Photochem. Photobiol., A*, 2017, **336**, 69–76.
- 11 M. R. Asabe and V. P. Ubale, *International Journal of Engineering Science Invention*, 2013, **2**, 41–44.
- 12 N. M. Khusayfan, *Aust. J. Basic Appl. Sci.*, 2012, **6**, 329–336.
- 13 V. M. Nikale, N. S. Gaikwad, K. Y. Rajpure and C. H. Bhosale, *Mater. Chem. Phys.*, 2003, **78**, 363–366.
- 14 T. Mahalingam, S. Thanikaikarasan, R. Chandramohan, K. Chung, J. P. Chu, S. Velumani and J.-K. Rhee, *J. Mater. Sci. Eng. B*, 2010, **174**, 236–241.
- 15 S. D. Dhruv, T. Dudharejiya, S. A. Sharko, A. I. Serokurova, N. N. Novitskii, D. L. Goroshko, R. Banerjee, J. Jangale, P. Rayani, V. Solanki, M. P. Deshpande, J. H. Markna, B. Kataria and D. K. Dhruv, *Appl. Phys. A*, 2025, **131**, 663.
- 16 S. D. Dhruv, S. A. Sharko, A. I. Serokurova, N. N. Novitskii, D. L. Goroshko, P. Rayani, J. Jangale, N. Agrawal, V. Solanki, J. H. Markna, B. Kataria and D. K. Dhruv, *RSC Adv.*, 2025, **15**, 14859–14875.
- 17 S. D. Dhruv, T. Dudharejiya, S. A. Sharko, A. I. Serokurova, N. N. Novitskii, D. L. Goroshko, P. Rayani, J. Jangale, V. Solanki, P. B. Patel, U. B. Trivedi, J. H. Markna, B. Kataria and D. K. Dhruv, *RSC Adv.*, 2025, **15**, 26353–26361.
- 18 H. Hahn, G. Frank, W. Klingler, A. D. Störger and G. Störger, *Z. Anorg. Allg. Chem.*, 1955, **279**, 241–270.
- 19 D. K. Dhruv, B. H. Patel, S. D. Dhruv, P. B. Patel, U. B. Trivedi and N. Agrawal, *Mater. Today: Proc.*, 2023, S2214785323002638.
- 20 S. D. Dhruv, J. Kolte, P. Solanki, M. P. Deshpande, V. Solanki, J. Tailor, N. Agrawal, V. A. Patel, J. H. Markna, B. Kataria and D. K. Dhruv, *RSC Adv.*, 2024, **14**, 15455–15467.
- 21 M. A. M. Seyam, G. F. Salem and S. N. A. Aziz, *Egypt. J. Solid.*, 2011, **34**, 88–98.
- 22 B. D. Cullity and S. R. Stock, *Elements of X-Ray Diffraction*, Prentice Hall, Upper Saddle River, NJ, 3rd edn, 2001.
- 23 R. Suresh, V. Ponnuswamy and R. Mariappan, *Appl. Surf. Sci.*, 2013, **273**, 457–464.
- 24 A. El-Habib, M. Addou, A. Aouni, M. Diani, J. Zimou and H. Bakkali, *Materialia*, 2021, **18**, 101143.
- 25 B. Parveen, Mahmood-ul-Hassan, Z. Khalid, S. Riaz and S. Naseem, *J. Appl. Res. Technol.*, 2017, **15**, 132–139.
- 26 V. M. Nikale, S. S. Shinde, A. R. Babar, C. H. Bhosale and K. Y. Rajpure, *Sol. Energy*, 2011, **85**, 1336–1342.
- 27 A. A. Attia, H. A. M. Ali, G. F. Salem, M. I. Ismail and F. F. Al-Harbi, *Opt. Mater.*, 2017, **66**, 480–486.
- 28 D. K. Dhruv, B. H. Patel, N. Agrawal, R. Banerjee, S. D. Dhruv, P. B. Patel and V. Patel, *J. Mater. Sci.: Mater. Electron.*, 2022, **33**, 24003–24015.
- 29 D. K. Dhruv, B. H. Patel and D. Lakshminarayana, *Mater. Res. Innovations*, 2016, **20**, 285–292.
- 30 P. Singh and N. M. Ravindra, *Sol. Energy Mater. Sol. Cells*, 2012, **101**, 36–45.
- 31 A. D. Dhass, Y. Prakash and K. C. Ramya, *Mater. Today: Proc.*, 2020, **33**, 732–735.
- 32 A. Seçkin and H. Koralay, *ACS Omega*, 2025, **10**, 6520–6533.
- 33 O. Madelung, *Semiconductors: Data Handbook*, Springer Berlin Heidelberg, Berlin, Heidelberg, 2004.
- 34 L. S. Koval, E. K. Arushanov and S. I. Radautsan, *Phys. Status Solidi A*, 1972, **9**, K73–K75.
- 35 A. Priyambada, A. Mohanty and P. Parida, *Mater. Today Commun.*, 2023, **37**, 107338.
- 36 H. Neumann, W. Kissinger and F. Lévy, *Cryst. Res. Technol.*, 1990, **25**, 1189–1193.
- 37 R. D. Jones, *Hybrid Circuit Design and Manufacture*, CRC Press, 1st edn, 2020.
- 38 Ç. Bilkan and Ş. Altındal, *J. Alloys Compd.*, 2017, **708**, 464–469.
- 39 Z. H. Mahmoud, Y. Ajaj, A. M. Hussein, H. N. K. Al-Salman, M. A. Mustafa, E. H. Kadhum, S. Abdullaev, S. A. Khuder, G. K. Ghadir, S. M. Hameed, K. Muzammil, S. Islam and E. Kianfar, *Int. J. Biol. Macromol.*, 2024, **267**, 131465.

