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Ion-selective uranium capture from seawater by UiO-66 metal–organic framework modified with amidoxime groups†

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Seawater is a valuable source of uranium (U) resources, and harnessing it effectively can play a crucial role in promoting nuclear energy. However, current polymer-based adsorbents typically exhibit slow adsorption rates and insufficient selectivity. In this work, the efficient uranyl ion (UO_2^{2+}) adsorbent (UiO-66-AO) was obtained from UiO-66-NH₂ through a simple and mild approach post synthesis modification (PSM). During the PSM process, not only the octahedral morphology (particle size ~200 nm) but also the crystal structure of UiO-66-NH₂ was well maintained. The integration of amidoxime groups (AO) improved the selectivity and adsorption performance towards UO_2^{2+} . According to the Langmuir model, the maximum adsorption capacity (q_{max}) of UiO-66-AO was 413.2 mg g⁻¹. In the competitive ion solution containing vanadium(v), iron (Fe), magnesium (Mg), calcium (Ca), etc., UiO-66-AO exhibited a much higher UO_2^{2+} adsorption capacity of 12.3 mg g⁻¹ than UiO-66-NH₂ (0.9 mg g⁻¹). Furthermore, UiO-66-AO achieved equilibrium for UO_2^{2+} adsorption in natural seawater within 3 days, and the adsorption capacity was 3.0 mg g⁻¹. Finally, near-edge X-ray absorption fine structure (NEXAFS) and X-ray photoelectron spectroscopy (XPS) demonstrated the chelation effect of AO on U, proving that the PSM successfully achieved performance enhancement. It is hoped that the simple and mild AO modification approach will provide new insights for the preparation of rapid, highly selective, and efficient UO_2^{2+} adsorbents.

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1. Introduction

With the rapid increase in global energy demand, nuclear energy has gained significant attention for its high energy density and zero greenhouse gas emissions.^{1,2} In the long run, the demand for uranium (U) shows a continuous growth trend. However, it is estimated that global conventional U resources are only used by humans for about 100 years.³ Fortunately, the reserves of U in the ocean are abundant, approximately 1000 times that of land, which can compensate for the environmental problems and limited reserves of traditional mining of U from land.^{4,5} Thanks to its simplicity of operation, high efficiency, and economic feasibility, the adsorption method remains the mainstream approach for uranium extraction from seawater (UES).

In recent years, metal–organic frameworks (MOFs) constructed from metal ions or clusters and organic ligands, such as UiO-66-3C4N, UiO-66-neomycin, MIL-101, have attracted

significant attention for uranyl ion (UO_2^{2+}) adsorption due to their high specific surface area, porosity, and structural adjustability.^{6–9} However, the concentration of UO_2^{2+} in the ocean is extremely low (≈ 3.3 ppb) and is accompanied by various competing ions.^{10,11} Therefore, effective UES necessitates enhancing the selectivity of MOFs for UO_2^{2+} and improving their affinity under trace UO_2^{2+} conditions.

Furthermore, achieving MOF adsorbents with targeted performance through simple and efficient methods remains a considerable challenge.^{12–14} In a recent study, Ghosh and coworkers reported an ionic mesoporous material i-MZIF(90)50 derived from UiO-66-NH₂ and ZIF-90 MOFs through precise morphology engineering, which benchmark record of 28.2 mg g⁻¹ UO_2^{2+} uptake from natural non-spiked seawater.¹⁵ Recent advancements in the synthesis and functionalization of MOFs have highlighted post-synthetic modification (PSM) as an effective and versatile strategy to enhancing the performance of MOFs. This technique involves reactions conducted after the initial synthesis of the frameworks, allowing for tailored modifications.^{16–19} PSM allows for modifications to the surface environment of MOFs while preserving the original bridging coordination and topological structure.^{20,21} These modifications can lead to the emergence of new chemical and physical properties, allowing for adjustments in the pores, surface characteristics, stability, hydrophobicity, adsorption capacity, catalytic activity,

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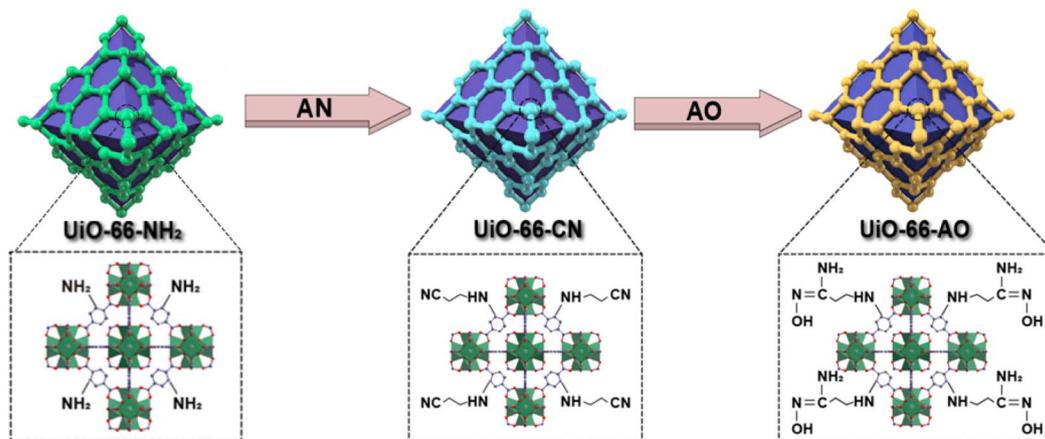


Fig. 1 Illustration of the synthesis of UiO-66-AO via PSM.

luminescence, and magnetic properties of MOFs, thus providing new opportunities for applications.^{22–25} The amidoxime group (AO) is currently one of the most commonly utilized functional groups for UES.²⁶ By introducing AO into MOFs, the selectivity and adsorption efficiency for U can be significantly improved.

To address the low selectivity and complex preparation processes of current MOFs adsorbents for UES. This study selected UiO-66-NH₂ as the initial MOFs due to its high framework stability in water. Acetic acid was used as a modulator to affect the crystallinity of MOFs. Zr₆ MOFs are ideally 12-coordinated. As acetic acid added, Zr₆ MOFs with reduced coordination (8- and 6-coordinated) can be considered as MOFs with defect. Indeed, the modulator facilitates the formation of defects in MOFs, which have an important influence on the stability, reactivity, porosity and thermomechanical behavior of

MOFs.^{16,27} Subsequently, AO was integrated through PSM to obtain UiO-66-AO material (Fig. 1). During the PSM process, the framework structure and morphology of MOFs were well maintained. UiO-66-AO exhibited good selectivity for UO₂²⁺ under trace conditions and rapid adsorption rate in seawater, providing reliable insights for MOFs modification and presenting promising materials for UES.

2. Results and discussion

2.1 Preparation of UiO-66-AO

The desired adsorbent, UiO-66-AO, was synthesized following the route depicted in Fig. 2a. UiO-66-NH₂ was synthesized following established methods from previous studies.²⁸ UiO-66-CN was synthesized by PSM in the presence of acrylonitrile (AN)

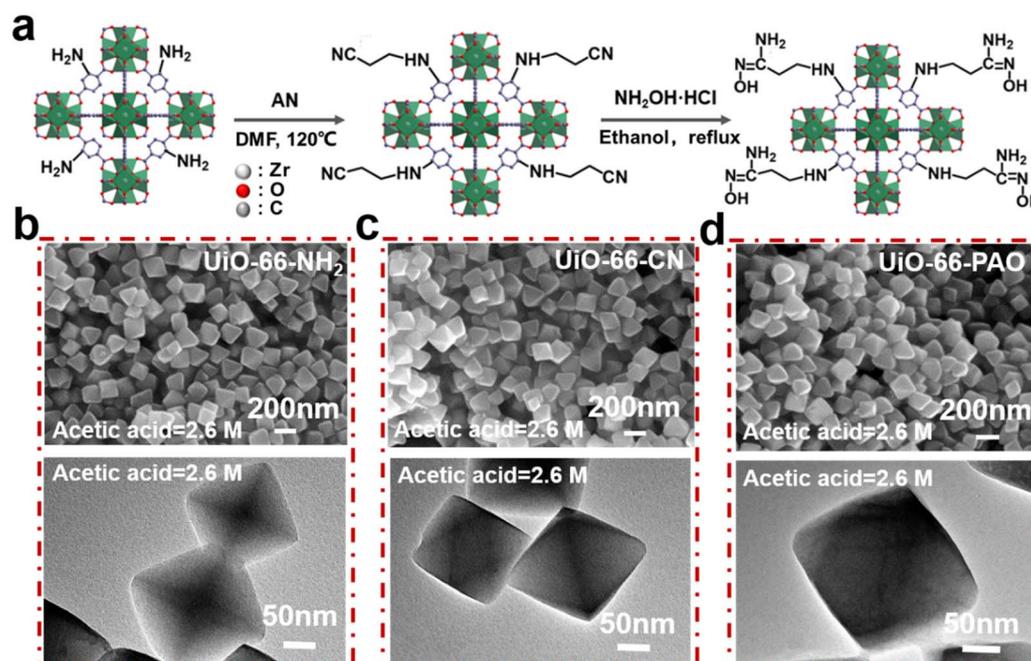


Fig. 2 (a) UiO-66-AO synthesis process. SEM and TEM images of (b) UiO-66-NH₂, (c) UiO-66-CN, and (d) UiO-66-AO.

through a Micheal addition reaction.²⁹ The final adsorbent, UiO-66-AO, was synthesized by refluxing UiO-66-CN with hydroxylamine hydrochloride in ethanol for 24 hours, with triethylamine serving as the base. More detailed synthesis information can be found in the supporting literature.

Acetic acid, as a modulator, can effectively regulate the morphology of MOFs particles, in turn, affect the UO_2^{2+} adsorption capacity.¹⁶ The effect of acetic acid concentration on the synthesis of UiO-66-NH₂ was studied firstly. As shown by SEM in Fig. S1a₀†, when without the acetic acid and the concentration of $\text{ZrCl}_4 : \text{BDC-NH}_2$ was 1.0 : 1.5, UiO-66-NH₂ clusters were successfully formed. With the concentration of acetic acid increased (1.0, 1.7, 2.6, and 4.1 M), the clusters became increasingly dispersed (Fig. S1a₁–a₄†). When the concentration of acetic acid was 2.6 M, the octahedral particles of UiO-66-NH₂ were more uniform with particle sizes around 200 nm (Fig. 2b and S1a₃†). As the concentration of acetic acid further increased, at a concentration of 4.1 M (Fig. S1a_b†), although the octahedral morphology continued to be maintained, significant defects and damage appeared in the structure. After PSM process with UiO-66-NH₂, the UiO-66-CN clusters were reacted, and it could be found their aggregate morphology maintained (Fig. S1b₀–b₄†). The morphology of UiO-66-AO remained intact after the PSM process with UiO-66-CN (Fig. S1c₀–c₄†). Fig. 2c and d show that the PSM process had minimal impact on the crystal structure and morphology of the original UiO-66-NH₂.

When the concentration of $\text{ZrCl}_4 : \text{BDC-NH}_2$ was 1.0 : 1.0, the result was consistent with the observation described above (Fig. S2†). With higher concentrations of acetic acid, the MOFs particles became more dispersed, and the octahedral shape became more pronounced. However, at excessively high acetic acid concentrations, defects appeared in the MOFs particles,

likely due to acetic acid's effect on the coordination between metal centers and ligands.³⁰ In addition, the PSM process had little influence on the structural integrity of the MOFs.

2.2 Characterization of materials

A comprehensive study was carried out on the physicochemical properties of UiO-66-NH₂ during the PSM. As shown in Fourier transform infrared spectroscopy (FT-IR) spectrum (Fig. 3a), UiO-66-NH₂ had two signs appeared at 3400–3500 cm^{-1} , corresponding to the stretching vibrations of –NH₂, which bonded to the benzene ring. Following the Michael addition of the cyano group, new stretching vibrations of –C≡N (–CN) appeared at 2247 cm^{-1} , confirming the successful grafting of AN onto UiO-66-NH₂. Ultimately, the disappearance of the –CN group, along with the emergence of stretching vibrations for –C=N– (1642 cm^{-1}) and N–O (916 cm^{-1}), confirmed the conversion of –CN groups to AO groups through amidoximation.³¹ Analysis of the full X-ray photoelectron spectroscopy (XPS) in Fig. 3b revealed an increase of the N 1s peak (399 eV) after PSM, while the Zr 3d peak (284 eV) remained largely unchanged, which further proved that the –CN and AO groups were successfully obtained.

The thermal stability of MOFs during PSM was examined using thermogravimetric analysis (TG). As shown in Fig. 3c, the thermal stability of MOFs during PSM varied with different functionalities, as indicated by the TG results. The weight loss of MOFs below 120 °C was caused by solvent evaporation.³¹ The functional groups of UiO-66-CN decomposed after ~300 °C. After amidoximation, the apparent decomposition temperature of AO group was ~150 °C, which was consistent with literature reports. All TG curves indicated that the apparent starting temperature for MOFs decomposition to ZrO_2 was ~525 °C.³² As

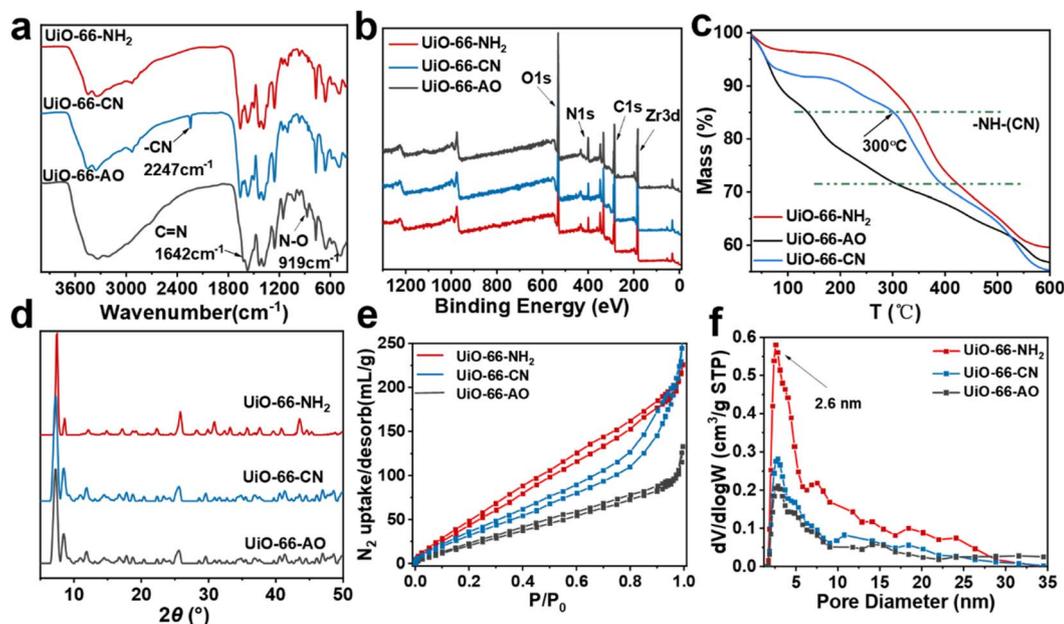


Fig. 3 Characterization of UiO-66-NH₂ before and after modification. (a) FT-IR spectra, (b) XPS spectra, (c) TG curves, (d) PXRD patterns, (e) N₂ adsorption/desorption isotherms, and (f) pore size distributions of UiO-66-NH₂, UiO-66-CN, and UiO-66-AO.



shown in Fig. 3d, the powder X-ray diffraction (PXRD) spectrum of MOFs during the PSM were almost consistent with the peaks of the original UiO-66-NH₂, indicating that the crystal structure was formed and maintained in the process.³³ It was demonstrated that UiO-66-AO was successfully prepared, and the combination of SEM and TEM (Fig. 2b–d) further showed that the influence of PSM on the crystal structure of the MOFs was not significant. N₂ adsorption/desorption isotherms were investigated as shown in Fig. 3e. Apparently, the three isotherms were the combination of type I and type IV isotherms curves according to the IUPAC classification, indicating the presence of microporous and mesoporous in the structures.³⁴ Using the Brunauer–Emmett–Teller method, the specific surface areas were measured as 255.82, 168.75, and 109.66 m² g⁻¹, respectively (correspond to UiO-66-NH₂, UiO-66-CN, and UiO-66-AO). The surface area of MOFs gradually decreased with the PSM process, mainly due to the larger functional groups appeared in the ligand. Micropores contribute a substantial specific surface area and pore volume, boosting the availability of active sites. Mesopores and macropores aid in the quick diffusion of ions and efficient transfer of substances.³⁵ The pore size was primarily centered around ~2.6 nm (Fig. 3f), supporting effective diffusion of UO₂²⁺.³⁶ The number of pores concentrated at 2.6 nm gradually decreased, this was also attributed to pore blockage caused by the large functional groups introduced during the PSM process.

2.3 Uranium adsorption experiments

To optimize the adsorption material, the effects of ligand concentration and acetic acid concentration on UO₂²⁺ adsorption were examined. As shown in Fig. S5a,† when the

concentration of acetic acid was constant, the effect of the concentration of ligand (molar ratio of ZrCl₄ : BDC-NH₂ = 1.0 : 1.0 or 1.0 : 1.5) on UO₂²⁺ adsorption was not obvious. However, while the concentration of ligand was constant, as the concentration of acetic acid increased, the adsorption capacity also increased. As the acetic acid concentration was 2.6 M, the adsorption capacity reached its maximum value (10.9 mg g⁻¹). When the molar ratio was increased further, the adsorption capacity decreased because the acetic acid disrupted the MOFs structure (Fig. S1a₄–c₄†). Acetic acid can regulate adsorption, which may be due to its modulation of the crystallinity of MOFs, affecting the active sites concentration of hydroxo, aquo or monocarboxylic acid on Zr₆ coordination.¹⁶ UiO-66-AO prepared with ZrCl₄ : BDC-NH₂ = 1.0 : 1.5 and acetic acid concentration 2.6 M was used as the sample for subsequent adsorption.

Secondly, optimization experiments were conducted on the solid–liquid ratio of the adsorption system. As shown in Fig. 4a, the adsorption capacity gradually decreased (from 97.0 to 6.1 mg g⁻¹) with the m V⁻¹ increased (from 0.002 to 0.05), and the removal rate of UO₂²⁺ in the solution gradually increased (from 58.8% to 92.2%). For subsequent adsorption experiments, 0.02 mg mL⁻¹ was chosen for adsorption testing, as its removal rate was moderate and the transition was significant.

The forms of UO₂²⁺ species in solution are highly influenced by the pH level of the solution. Protons not only influence U speciation but also alter the surface charge of the sorbent.^{37,38} UiO-66-AO could maintain a good octahedral morphology at pH = 3.0–9.0. When pH = 2.0 and 10.0, obvious defects were appeared on the surface of the sample (Fig. S3†). Additionally, to evaluate the impact of pH on UO₂²⁺ adsorption and identify the ideal value, adsorption experiments were carried out with pH range from 3.0 to 9.0. As indicated in Fig. 4b, the adsorption

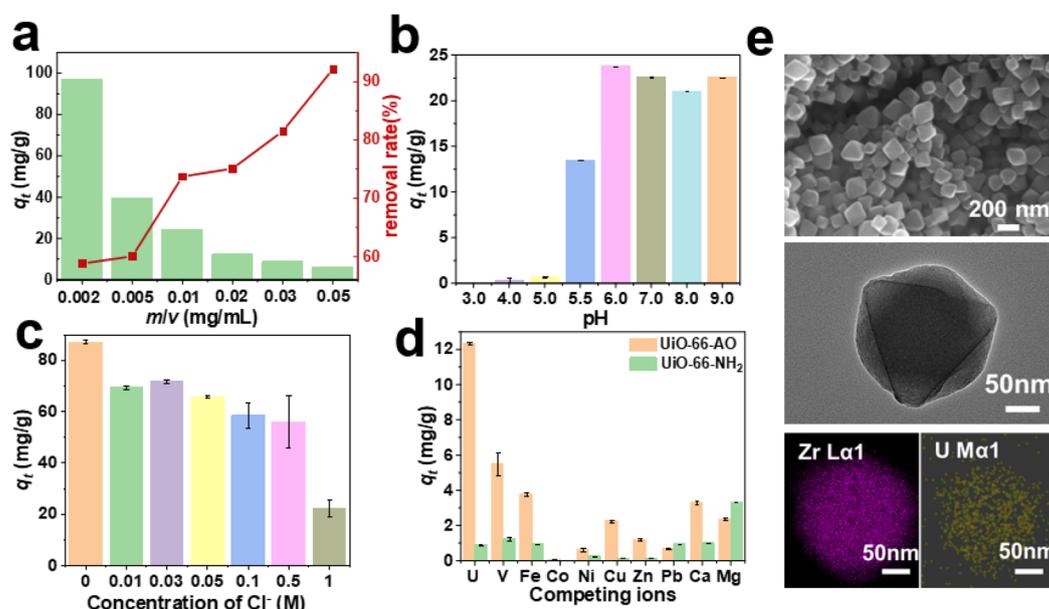


Fig. 4 (a) Effect of the mass-to-volume ratio on adsorption performance. (b) Influence of pH on UO₂²⁺ adsorption ($c_0 = 500$ ppb, $t = 24$ h), (c) the effect of anion Cl⁻ concentration on UO₂²⁺ adsorption ($c_0 = 2.0$ ppm, $t = 24$ h). (d) Adsorption in simulated seawater (ion concentrations provided in Table S1,† $t = 24$ h). (e) SEM and TEM of UiO-66-AO after the adsorption of UO₂²⁺ in seawater, and its Zr, and U elemental mapping photographs.

capacity of UiO-66-AO was below 1.0 mg g^{-1} when the pH was ≤ 5.0 , but it peaked at 23.8 mg g^{-1} at pH 6.0. With the increase of pH ($\text{pH} > 6.0$), the adsorption capacity decreased slightly but was still more than 21.0 mg g^{-1} . This characteristic indicates the potential of UiO-66-AO for UES.¹¹

As depicted in Fig. 4c, the influence of counter anions, specifically chloride ions (Cl^-), on UO_2^{2+} adsorption was investigated. The concentration of Cl^- was adjusted by adding NaCl to the U solution. When there was no Cl^- in solution, the maximum adsorption capacity of UO_2^{2+} was 87.2 mg g^{-1} . When the Cl^- concentration was 0.01 and 0.05 M, the UO_2^{2+} adsorption capacity decreased slightly to 69.4 and 65.8 mg g^{-1} , respectively. When the Cl^- concentration $\geq 0.1 \text{ M}$, as the concentration of Cl^- ions increase, the adsorption capacity gradually decreased, and the UO_2^{2+} adsorption capacity dropped to 22.3 mg g^{-1} when the Cl^- concentration was 1 M, which meant that the effect on the adsorption of UO_2^{2+} was weak when the Cl^- concentration was less than 0.05 M.

To study the impact of competing ions on UO_2^{2+} adsorption, experiments were conducted using simulated seawater. Competing metal ions were chosen according to prior studies, with their concentrations listed in Table S1.[†] As shown in Fig. 4d, the UO_2^{2+} adsorption capacity of UiO-66-AO reached 12.3 mg g^{-1} , greatly surpassing that of UiO-66-NH₂ at 0.9 mg g^{-1} . This highlights that introducing AO *via* the PSM method has significantly improved the UO_2^{2+} adsorption performance of the original UiO-66-NH₂. In addition, the adsorption capacity of UiO-66-AO for UO_2^{2+} was higher than that for V (5.5 mg g^{-1}), Fe (3.7 mg g^{-1}), Co (0.1 mg g^{-1}), Ni (0.6 mg g^{-1}), Cu (2.3 mg g^{-1}), Zn (1.2 mg g^{-1}), Pb (0.7 mg g^{-1}), Ca (3.3 mg g^{-1}), and Mg (2.3 mg g^{-1}), this indicates that UiO-66-AO demonstrates outstanding selectivity for U.⁴² Homogeneous distribution of

UO_2^{2+} on the surface of UiO-66-AO (U@UiO-66-AO) could be clearly observed after U adsorption in simulated seawater (Fig. S4[†]).

Building on the excellent UO_2^{2+} adsorption performance of UiO-66-AO observed above, further investigations were conducted to evaluate its ability to UES. As depicted in Fig. S5b,[†] UiO-66-AO was able to reach adsorption equilibrium for UO_2^{2+} in natural seawater within 3 days, demonstrating an adsorption capacity of 3.0 mg g^{-1} , exhibiting good adsorption performance and fast adsorption kinetics (Table S2[†]), which is related to the number of active sites provided by micropores and rapid ion diffusion promoted by mesopores.³⁵ In addition, the morphology of UiO-66 after adsorbing UO_2^{2+} with 16 days remained octahedral morphology (Fig. 4e), indicating good stability of its structure in seawater. Elemental mapping (Fig. 4e and S6b[†]) revealed the distribution of UO_2^{2+} on UiO-66-AO, demonstrating its strong ability to UES.

The adsorption kinetics of UO_2^{2+} on the UiO-66-AO at pH 6 is shown in Fig. 5a. A significant increase of the adsorption rate of UO_2^{2+} was observed within 400 min, and then a high-level of adsorption capacity (286.8 mg g^{-1}) was observed within 1100 min, which exhibited a faster adsorption rate than UiO-66-NH₂(AO) reported previously.²⁹ The adsorption kinetics of UO_2^{2+} on the UiO-66-AO was tested by the pseudo-first-order and pseudo-second-order kinetic models.³¹ And the data of the adsorption kinetics are shown in Table 1. UO_2^{2+} adsorption on UiO-66-AO is more accurately described by the pseudo-second-order model, showing a high correlation coefficient ($R^2 = 0.994$, Fig. 5b) compared to the pseudo-first-order model ($R^2 = 0.981$, Fig. 5c). This indicates that chemisorption likely serves as the main rate-determining step.⁴³

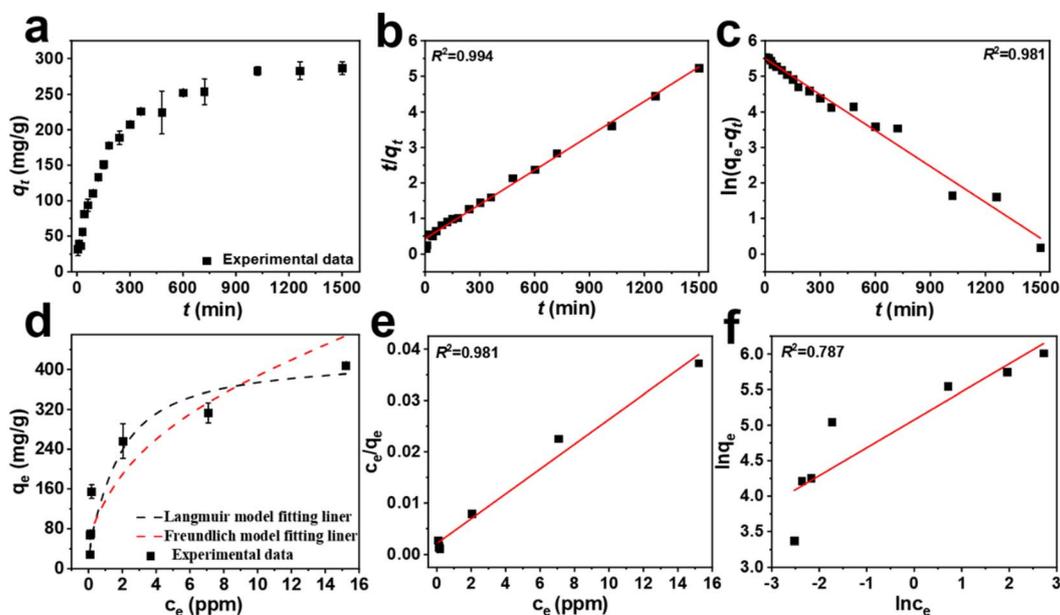


Fig. 5 (a) Impact of contact time on UO_2^{2+} adsorption ($c_0 = 8 \text{ ppm}$, $t = 24 \text{ h}$), (b) the pseudo-second-order and (c) the pseudo-first-order kinetic model linearized plots for UO_2^{2+} adsorption, (d) impact of UO_2^{2+} concentration on the adsorption ($c_0 = 0.1\text{--}20.0 \text{ ppm}$, $t = 24 \text{ h}$), (e and f) fitting results for the Langmuir and Freundlich isothermal adsorption models of UiO-66-AO.



Table 1 The adsorption kinetic fitting parameter

Concentration	Pseudo-second-order model			Pseudo-first-order model		
	R^2	q_e (mg g ⁻¹)	K_2 (g·mg ⁻¹ ·min ⁻¹)	R^2	q_e (mg g ⁻¹)	K_1 (g·mg ⁻¹ ·min ⁻¹)
8 ppm	0.994	309.6	2.44×10^{-5}	0.981	245.0	3.37×10^{-3}

Table 2 The adsorption isothermal model fitting parameters

Adsorbent	Langmuir model			Freundlich model		
	R^2	q_{\max} (mg g ⁻¹)	b (L mg ⁻¹)	R^2	k_F	n
2 mg	0.981	413.2	1.158	0.787	160.0	2.539

The influence of UO_2^{2+} initial concentration on adsorption was assessed (0.1–20.0 ppm). As illustrated in Fig. 5d, U adsorption continuously increased with higher initial concentrations. At equilibrium, the maximum sorption capacity reached 408.0 mg g⁻¹ under the experimental conditions. The adsorption isotherm data were modeled using the Langmuir and Freundlich isotherms (Fig. 5e and f).⁴⁴ R^2 of Langmuir model was 0.981, with a higher R^2 value than the Freundlich model ($R^2 = 0.787$, Table 2), the results suggest that adsorption occurs as a monolayer on the adsorbent's surface.³⁹ The estimated maximum adsorption capacity (q_{\max}) was 413.2 mg g⁻¹, which was significantly greater than the q_{\max} (134.1 mg g⁻¹) of UiO-66-NH-(AO). UiO-66-AO demonstrated a faster adsorption rate and greater adsorption capacity than UiO-66-NH-(AO), attributed to its larger specific surface area (the specific surface areas of UiO-66-NH-(AO) was 59.2 m² g⁻¹, which was much lower than UiO-66-AO), which offered more active sites and enhanced effective contact with UO_2^{2+} .³⁵

2.4 Adsorption mechanism of UiO-66-AO

The possible mechanism of UO_2^{2+} adsorption was investigated by XPS and NEXAFS. Through the full spectrum of XPS (Fig. 6a), there was a new sign of U 4f (381.1 eV) appeared in U@UiO-66-AO, this indicates that UiO-66-AO effectively achieved chemical adsorption of uranium (Fig. S6a†). The O 1s spectrum of UiO-66-AO (Fig. 6b) showed three signs, which were -OH (529.6 eV), C=O (531.1 eV), and N-O (531.8 eV). After adsorbing UO_2^{2+} , the peak of -OH shifted towards lower binding energy fields, while the peaks of N-O shifted towards higher binding energy fields. Additionally, the sign intensities of C=O and -OH showed a marked increase, while the intensity of the N-O sign decreased. Likely due to UO_2^{2+} adsorption by the AO groups in the MOFs. Analysis of the N 1s spectra of UiO-66-AO and U@UiO-66-AO (Fig. 6c) revealed two distinct characteristic peaks in the N 1s spectra of UiO-66-AO: C=N at 399.0 eV and C-N at 400.6 eV. Upon UO_2^{2+} adsorption, the C=N and C-N peaks shifted to lower binding energies, and their intensities decreased. This change was attributed to the interaction between the AO and the synergistic -NH- groups in the ligands.^{45,46} Fig. 6d showed the C 1s spectra of UiO-66-AO and U@UiO-66-AO. The C 1s spectrum of UiO-66-AO was deconvoluted into three bands at 283.6, 284.9, and 287.4 eV, corresponding to C-C, C-O/N, and C=O, respectively. After UO_2^{2+} adsorption, the C-O/N peak shifted to a higher binding energy,

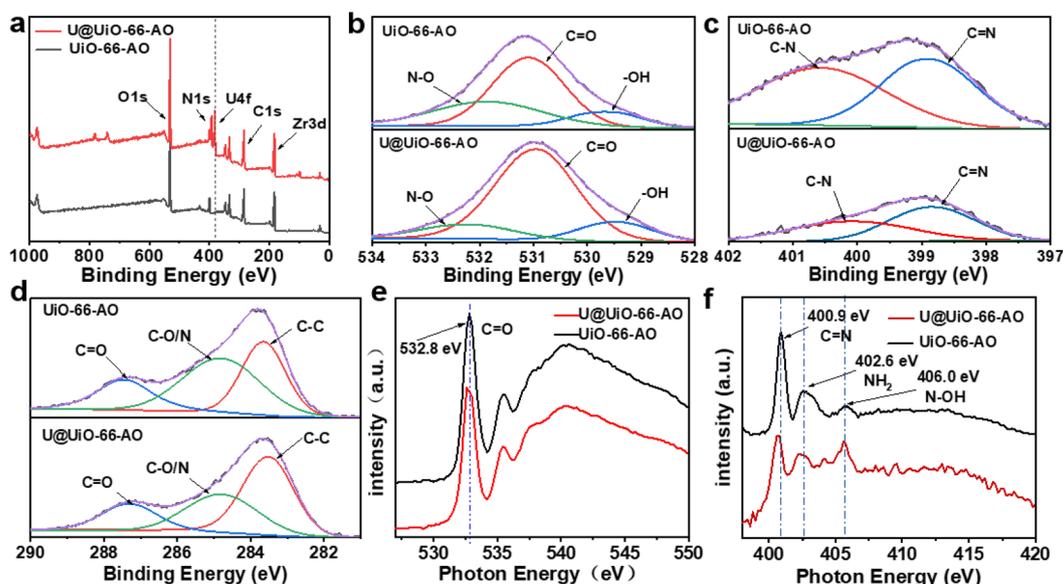


Fig. 6 (a) XPS spectra, (b) O 1s spectra, (c) N 1s spectra, (d) C 1s spectra and (e) O k -edge of NEXAFS and (f) N k -edge of NEXAFS of UiO-66-AO and U@UiO-66-AO.



and the intensities of C=O and C–O/N peaks decreased. This indicates that nitrogen- and oxygen-containing functional groups played a key role in the effective removal of UO_2^{2+} . The O *k*-edge NEXAFS of UiO-66-AO and U@UiO-66-AO was analyzed. As shown in Fig. 6e, the peak of UiO-66-AO at 532.8 eV corresponded to the O of sp^2 in C=O.⁴⁷ The π^* transition after UO_2^{2+} adsorption occurred at 400.9 and 402.6 eV, which shifted towards lower photon energy and may be related to the C=N and NH_2 in the AO (Fig. 6f).^{48,49} The coordination effect between AO and UO_2^{2+} was more fitted with a cooperative chelating model.^{10,49} The characterization results confirm that the high adsorption capacity of UiO-66-AO for U is primarily due to the efficient capture of UO_2^{2+} by the AO introduced through PSM, along with the synergistic coordination effects of the MOFs ligands.

3. Conclusions

A novel MOFs, UiO-66-AO was synthesized by grafting AO onto UiO-66- NH_2 via the PSM method. The octahedral morphology and crystal structure of the original MOFs UiO-66- NH_2 were maintained during the PSM process. But the surface environment of MOFs changed, AO was effectively grafted onto the surface of MOFs, which was conducive to effective contact with UO_2^{2+} , resulting in UiO-66-AO exhibiting strong adsorption capabilities and selective uptake of UO_2^{2+} . Adsorption equilibrium was reached within 1100 minutes. The maximum adsorption capacity (q_{max}) attained 413.2 mg g^{-1} . In simulated seawater, the adsorption capacity was 12.3 mg g^{-1} , much higher than that of the original UiO-66- NH_2 (0.9 mg g^{-1}), demonstrating strong affinity and selectivity for UO_2^{2+} . UiO-66-AO could reach adsorption equilibrium within 3 days in natural seawater, with a fast adsorption rate and an adsorption capacity of 3.0 mg g^{-1} . In addition, XPS and NEXAFS confirmed the chelation between AO and uranium, as well as the synergistic effect of ligands, indicating that the simple PSM method successfully improved the adsorption performance of original MOFs UiO-66- NH_2 . This study provides a feasible strategy for MOFs modification, and demonstrates the potential of MOFs-based materials for UES.

4. Materials and method

4.1 Reagents and materials

All chemicals were purchased from Sinopharm chemical reagent company. (Shanghai, China) and used without purification.

4.2 Preparation of UiO-66-AO

UiO-66- NH_2 was synthesized according to the previous literature with minor modifications.²⁸ UiO-66-CN was obtained by adding 0.60 g UiO-66- NH_2 into the mixed solution of 15 mL AN and 15 mL DMF. The mixture was stirred and heated at 120 °C for 12 h. Then solids were centrifuged and washed with DMF and ethanol. A certain amount of UiO-66-CN (typically 0.20 g) was added into the mixture of 1.29 g $\text{NH}_2\text{OH}\cdot\text{HCl}$, 1.88 g

triethylamine and 50 mL ethanol (which was dissolved and mixed in advance). Refluxing for 24 h, the solids were centrifuged and washed by ethanol, and the UiO-66-AO was obtained.

4.3 Batch adsorption experiments

Typically, the ratio of adsorbent mass to solution volume (m V^{-1}) was 0.02 mg mL^{-1} , an amount of the adsorbent (2.0 mg) was added to 100 mL UO_2^{2+} solution of given concentration and pH value in a plastic bottle. The pH was adjusted by 0.1 M or 1 M HCl and NaOH solution. After being stirred at 100 rpm and 25 °C for 24 h, the sample was taken and filtered by 0.2 μm nylon membrane filter. Adsorption in simulated seawater was carried out according to previous study.^{10,26,29} Seawater was collected from Qingdao, China and adsorbed under laboratory conditions. UiO-66-AO (37 mg) was stirred in 18.66 L seawater at room temperature for 16 days.

Data availability

All data are contained within the manuscript and are available upon request.

Author contributions

Data curation and writing – original draft: L. M.; formal analysis: L. M., C. H.; visualization: C. H., P. S.; investigation: C. H., Y. W.; validation: H. M., H. S.; supervision: H. M., Q. L.; writing – review & editing: Q. L., Y. L.; funding acquisition: L. M., Y. L. All authors have read and agreed to the published version of the manuscript.

Conflicts of interest

The authors declare no conflicts of interests.

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References

- 1 W. Lin, *Presented in Part at Conference AMER CHEMICAL SOC1155*, 16TH ST, Washington, 2017, vol. 3.
- 2 M. Mirza, R. Abdulaziz, W. C. Maskell, S. Wilcock, A. H. Jones, S. Woodall, A. Jackson, P. R. Shearing and D. J. L. Brett, *Energy Environ. Sci.*, 2023, **16**, 952–982.
- 3 H. Lindner and E. Schneider, *Energy Econ.*, 2015, **49**, 9–22.
- 4 W. Cui, F. Li, R. Xu, C. Zhang, X. Chen, R. Yan, R. Liang and J. Qiu, *Angew. Chem., Int. Ed.*, 2020, **59**, 17684–17690.



- 5 J. Kim, C. Tsouris, Y. Oyola, C. J. Janke, R. T. Mayes, S. Dai, G. Gill, L. J. Kuo, J. Wood, K. Y. Choe, E. Schneider and H. Lindner, *Ind. Eng. Chem. Res.*, 2014, **53**, 6076–6083.
- 6 H. Liu, T. Fu and Y. Mao, *ACS Omega*, 2022, **7**, 14430–14456.
- 7 Y. Yuan, S. Feng, L. Feng, Q. Yu, T. Liu and N. Wang, *Angew. Chem., Int. Ed.*, 2020, **59**, 4262–4268.
- 8 J. Xiong, Y. Fan and F. Luo, *Dalton Trans.*, 2020, **49**, 12536–12546.
- 9 Y. Zeng, Y. Ni, S. Liu, J. Xu, A. Zhang, Y. Song, L. Yang, A. Pu, X. Li, L. Lv, L. Yuanli and F. Chi, *Microporous Mesoporous Mater.*, 2022, **343**, 112123–112130.
- 10 C. Huang, L. Xu, X. Xu, L. Ma, H. Bao, J. Liao, J. Wang, J. Han, G. Xu, D. Huang, B. Ye, H. Zhang, M. Wu, X. Zhao and H. Ma, *Chem. Eng. J.*, 2022, **443**, 136312–136321.
- 11 Y. Yuan, T. Liu, J. Xiao, Q. Yu, L. Feng, B. Niu, S. Feng, J. Zhang and N. Wang, *Nat. Commun.*, 2020, **11**, 1–8.
- 12 C. X. Yu, W. Jiang, M. Lei, M. R. Yao, X. Q. Sun, Y. Wang, W. Liu and L. L. Liu, *Small*, 2023, 2308910–2308921.
- 13 P. Sarawade, H. Tan and V. Polshettiwar, *ACS Sustain. Chem. Eng.*, 2013, **1**, 66–74.
- 14 Y. D. More, S. Mollick, S. Saurabh, S. Fajal, M. Tricarico, S. Dutta, M. M. Shirolkar, W. Mandal, J. Tan and S. K. Ghosh, *Small*, 2024, **20**, 2302014–2302023.
- 15 S. Mollick, S. Saurabh, Y. D. More, S. Fajal, M. M. Shirolkar, W. Mandal and S. K. Ghosh, *Energy Environ. Sci.*, 2022, **15**, 3462–3470.
- 16 S. M. Cohen, *J. Am. Chem. Soc.*, 2017, **139**, 2855–2863.
- 17 O. Karagiari, W. Bury, J. E. Mondloch, J. T. Hupp and O. K. Farha, *Angew. Chem., Int. Ed.*, 2014, **53**, 4530–4540.
- 18 C. K. Brozek and M. Dincă, *Chem. Soc. Rev.*, 2014, **43**, 5456–5467.
- 19 A. Tyagi, H. Sharma, A. K. Yadav, P. Singhal, G. Karmakar, D. Bhattacharyya and S. K. Jha, *Ind. Eng. Chem. Res.*, 2024, **63**, 10892–10902.
- 20 S. Mandal, S. Natarajan and P. Mani, *Adv. Funct. Mater.*, 2021, **31**, 2006291–2006313.
- 21 Z. Yin, S. Wan, J. Yang, M. Kurmoo and M. H. Zeng, *Coord. Chem. Rev.*, 2019, **378**, 500–512.
- 22 M. Zhang, T. Yang, Z. Wang, X. F. Ma and Y. Zhang, *Chem. Sci.*, 2017, **8**, 5356–5361.
- 23 S. Park, H. Song, N. Ko, C. Kim, K. Kim and E. Lee, *ACS Appl. Mater. Interfaces*, 2018, **10**, 33785–33789.
- 24 Z. Wang, K. K. Tanabe and S. M. Cohen, *Chem.–Eur. J.*, 2010, **16**, 212–217.
- 25 C. Bi, C. Zhang, C. Wang, L. Zhu, R. Zhu, L. Liu, Y. Wang, F. Ma and H. Dong, *Environ. Sci. Pollut. Res.*, 2024, **31**, 16554–16570.
- 26 X. Xu, H. Zhang, J. AO, L. Xu, X. Liu, X. Guo, J. Li, L. Zhang, Q. Li, X. Zhao, B. Ye, D. Wang, F. Shen and H. Ma, *Energy Environ. Sci.*, 2019, **12**, 1979–1988.
- 27 Z. Zhao, R. Lei, Y. Zhang, T. Cai and B. Han, *J. Mol. Liq.*, 2022, **367**, 120514–120522.
- 28 X. Shi, X. Lian, D. Yang, X. Hu, J. Zhang and X. H. Bu, *Dalton Trans.*, 2021, **50**, 17953–17959.
- 29 L. Ma, J. Gao, C. Huang, X. Xu, L. Xu, R. Ding, H. Bao, Z. Wang, G. Xu, Q. Li, P. Deng and H. Ma, *ACS Appl. Mater. Interfaces*, 2021, **13**, 57831–57840.
- 30 T. L. Ren, M. M. Wang, T. Z. Xiong, X. N. Zhang, Y. Peng, H. R. Wen and S. J. Liu, *Cryst. Growth Des.*, 2023, **24**, 696–702.
- 31 L. Chen, Z. Bai, L. Zhu, L. Zhang, Y. Cai, Y. Li, W. Liu, Y. Wang, L. Chen, J. Diwu, J. Wang, Z. Chai and S. Wang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 32446–32451.
- 32 J. Li, Y. Liu, X. Wang, G. Zhao, Y. Ai, B. Han, T. Wen, T. Hayat, A. Alsaedi and X. Wang, *Chem. Eng. J.*, 2017, **330**, 1012–1021.
- 33 J. Li, K. Tuo, C. Fan, G. Liu, S. Pu and Z. Li, *Small*, 2024, **20**, 2306545–2306555.
- 34 W. Liu, X. Dai, Z. Bai, Y. Wang, Z. Yang, L. Zhang, L. Xu, L. Chen, Y. Li, D. Gui, J. Diwu, J. Wang, R. Zhou, Z. Chai and S. Wang, *Environ. Sci. Technol.*, 2017, **51**, 3911–3921.
- 35 J. Kang, J. Min, S. I. Kim, S. W. Kim and J. H. Jang, *Mater. Today Energy*, 2020, **18**, 100502–100510.
- 36 B. E. Cowie, J. M. Purkis, J. Austin, J. B. Love and P. L. Arnold, *Chem. Rev.*, 2019, **119**, 10595–10637.
- 37 F. Endrizzi, C. J. Leggett and L. Rao, *Ind. Eng. Chem. Res.*, 2016, **55**, 4249–4256.
- 38 C. J. Leggett, F. Endrizzi and L. Rao, *Ind. Eng. Chem. Res.*, 2016, **55**, 4257–4263.
- 39 L. Ma, C. Huang, Y. Yao, M. Fu, F. Han, Q. Li, M. Wu, H. Zhang, L. Xu and H. Ma, *Sep. Purif. Technol.*, 2023, **314**, 123526–123538.
- 40 C. Huang, M. Fu, L. Ma, C. Mao, Y. Yao, X. Xu, L. Xu, J. Han, X. Xue, G. Xu, M. Wu, H. Shao and H. Ma, *Chem. Eng. J.*, 2023, **474**, 145718–145729.
- 41 P. Sun, L. Ma, J. Liao, J. Wang, C. Huang, L. Xu, M. Fu, J. Wang, Q. Zhou and H. Ma, *Sep. Purif. Technol.*, 2024, **337**, 126419–126428.
- 42 S. C. Liu, M. B. Wu, H. Ye, L. Liu, L. L. Ma and J. Yao, *Chem. Eng. J.*, 2021, **426**, 131378–131388.
- 43 C. A. Trickett, K. J. Gagnon, S. Lee, F. Gándara, H. Bürgi and O. M. Yaghi, *Angew. Chem., Int. Ed.*, 2015, **127**, 11314–11320.
- 44 R. M. Abdelhameed, M. Taha, H. Abdel-Gawad and H. E. Emam, *J. Mol. Struct.*, 2022, **1250**, 131914–131924.
- 45 J. Górka, R. T. Mayes, L. Baggetto, G. M. Veith and S. Dai, *J. Mater. Chem. A*, 2013, **1**, 3016–3026.
- 46 Y. Wang, Z. Gu, J. Yang, J. Liao, Y. Yang, N. Liu and J. Tang, *Appl. Surf. Sci.*, 2014, **320**, 10–20.
- 47 R. G. Ryan, A. Stacey, K. M. O'Donnell, T. Ohshima, B. C. Johnson, L. C. L. Hollenberg, P. Mulvaney and D. A. Simpson, *ACS Appl. Mater. Interfaces*, 2018, **10**, 13143–13149.
- 48 J. Melke, B. Peter, A. Habereeder, J. Ziegler, C. Fasel, A. Nefedov, H. Sezen, C. Woll, H. Ehrenberg and C. Roth, *ACS Appl. Mater. Interfaces*, 2016, **8**, 82–90.
- 49 X. Xu, C. Huang, Y. Wang, X. Chen, Z. Wang, J. Han, M. Wu, G. Liu, L. Li, L. Xu and H. Ma, *Chem. Eng. J.*, 2022, **430**, 133159–133170.

