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Understanding the activity origin and mechanisms of the oxygen reduction reaction on the tetramethyl metalloporphyrin/MoS₂ electrocatalyst†

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The efficiency of the oxygen reduction reaction (ORR) on the cathode plays a crucial role in determining the performance of proton exchange membrane fuel cells. Porphyrin, distinguished by its cost-effectiveness, eco-friendly nature, and efficient utilization of its metal, stands out as a promising candidate for a metal single-atom catalyst in fuel cell cathodes. The metal and support modifications significantly impact the porphyrin's ORR activity. Nevertheless, the effects of Ni, Co, and Fe metals in tetramethyl metalloporphyrin/MoS₂, named MeTMP/MoS₂, catalyst on the mechanisms and activity of the ORR remain unknown. This study elucidates the topic using van der Waals dispersion-corrected density functional theory (DFT) calculations and thermodynamic model. Results showed that the rate-limiting step is located at the first and second hydrogenation steps in the associative mechanisms for Ni and Co (Fe) substitutions, respectively. For the dissociative mechanisms, the dissociation of molecular oxygen to two oxygen atoms is the rate-determining step on all the NiTMP/MoS₂, CoTMP/MoS₂, and FeTMP/MoS₂ catalysts. The presence of the MoS₂ support significantly reduces the thermodynamic activation barrier of the ORR, and hence improves the ORR activity in the dissociative mechanisms. This activation barrier is 3.45, 0.92, and 1.82 eV for NiTMP/MoS₂, CoTMP/MoS₂, and FeTMP/MoS₂, which is much better compared to 4.85, 3.34, and 2.19 eV for NiTMP, CoTMP, and FeTMP, respectively. CoTMP/MoS₂ is the best candidate among the considered catalysts for the ORR. Furthermore, we provide a detailed explanation of the physical insights into the interaction between the ORR intermediates and the catalysts.

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1. Introduction

The ORR is a critical process in energy technologies, such as proton exchange membrane fuel cells and metal-air batteries.^{1,2} However, the sluggish kinetics of the ORR on the cathodes hinder the efficiency of these devices. Therefore, finding efficient and sustainable ORR catalysts is crucial for advancing energy technologies. Metalloporphyrins, incorporating a non-

precious metal atom in the centre of the porphyrin ring, have emerged as promising alternatives to traditional noble metal catalysts.^{3–11} The metal centre, often transition metals such as iron, cobalt, nickel, and manganese, serves as an active site for catalysing the ORR.^{10,11} A unique advantage of metal porphyrins lies in their ability to host a single metal atom as the catalytic site, minimizing metal usage and maximizing the catalytic efficiency. Besides, the reactivity of the metal porphyrins toward ORR is highly dependent on the substituents of the porphyrin ring functional groups, as evidenced by experimental^{12–14} and theoretical investigations.^{15–17} Our previous study showed that substituting tetramethyl (–CH₃), amino (–NH₂), and carboxyl (–COOH) functional groups at the *meso*-positions of the iron porphyrin can alter the electron density distribution within the ring, affecting the ORR mechanisms and activity of the iron porphyrin.¹⁵ Among the functional groups studied, the amino and methyl groups exhibited high oxygen reduction activity due to lowering the activation energy of the rate-determining steps.¹⁵

Because of the low stability of pure metalloporphyrins, they are typically coated onto electrodes made of various materials,

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such as glassy carbon and graphite for the ORR.^{7–10,17–19} Porphyrins supported on carbon substrates were found to significantly enhance the ORR catalytic activity.^{18,19} Schilling *et al.* also showed that metalloporphyrin membranes integrated into carbon nanotubes offered superior electrical conductivity, and significantly improved the ORR activity.¹⁸ Electrode materials with highly porous structures, such as metal–organic frameworks integrated with metal porphyrin linkers, also exhibited selective catalytic activity for the ORR *via* the 4-electron mechanism.¹⁹ Although carbon materials are commonly used as supports for the ORR catalysts, they are corroded under electrochemical operating conditions of the ORR. Therefore, they weaken the long-term stability of the catalysts.^{20,21}

Recently, transition metal dichalcogenides such as MoS₂, a class of carbon-free materials, have attracted much attention as support materials for electrocatalysts due to the high flexibility and stability of their structure.^{22–26} For example, under appropriate conditions, MoS₂ offered high stability and a large surface area for dispersing catalysts.²⁷ The literature revealed that the layered structure of MoS₂ provides well-defined active sites for anchoring precious metal catalysts, such as platinum for electro-catalytic water splitting.²⁸ Additionally, the MoS₂-based materials have been demonstrated as alternative catalysts for the hydrogen evolution reaction (HER) and the ORR.^{22,23,29} The N-doped MoS₂ supported on carbon catalysts also showed a dramatic improvement in their ORR activity and stability for microbial fuel cells.³⁰ Recently, iron porphyrins reinforced on the MoS₂ support showed good oxygen reduction activity.²⁴ However, no research is available on the ORR mechanisms and activity on the metalloporphyrin/MoS₂ electro-catalysts, particularly, for the MeTMP/MoS₂ catalyst with Me = Ni, Co, Fe; and TMP = tetramethyl porphyrin.

This study clarifies the effects of metal substitutions in the metalloporphyrin/MoS₂ catalyst on the mechanisms and activity of the ORR using the van der Waals DFT calculations and the thermodynamic model. The research focuses on iron-series elements (Fe, Co, Ni) because they are earth-abundant, and their compounds with porphyrins have shown good ORR activities, as reported in previous studies.^{3,7–13} The results of this work will be useful for designing a rational electrocatalyst for ORR.

2. Computational details

This study utilizes the metallic phase of MoS₂ with the distorted octahedral coordinate structure, denoted as 1T'-MoS₂, because of the following reasons: (1) the semiconducting phase 2H-MoS₂ undergoes a phase transition naturally or by heat treatments, and hydrothermal synthesis into the metallic phase 1T'-MoS₂.^{29,31} (2) The metallic phase has a higher electrical conductivity; therefore, it is beneficial for making an electrode with a better electrical conductivity and therefore enhances catalytic performance.³² (3) the recent study showed that the metallic phase 1T'-MoS₂ exhibited significantly higher activity for the electrocatalytic HER in an acidic medium than that of 2H-MoS₂.³³ Based on the primitive unit cell of the monolayer structure of the 1T'-MoS₂ phase,³⁴ we built a supercell in the

slab model by repeating 3 and 5 times along the *a* and *b* unit vectors or the *x* and *y* axes, as shown in Fig. S1† (top view) of ESI, respectively. The monolayer has one atomic layer of Mo atoms, sandwiched between two atomic layers of S atoms. Each atomic layer has thirty atoms. The optimized structure of the supercell by our DFT calculations has the Mo–Mo distance of 3.17, 3.81, and 2.77 Å along the *y*-axis and two diagonal axes (Fig. S1†), respectively. The free space of the supercell along the *c* direction (see the side view in Fig. S1†) is 17 Å, which is large enough for the adsorption of the metalloporphyrin molecule and the ORR intermediates on the 1T'-MoS₂ substrate without the crossing interaction between the supercell images.

Our previous study also showed that the *meso*-tetra-methyl (–CH₃) functional group is the best substitution among the considered substituents.¹⁵ The metal centre (Me) of the metalloporphyrin (MeTMP) is replaced sequentially with Ni, Co, and Fe. Here, we used the oxidation state +2 for Fe, Co, and Ni, which agrees with the literature.^{3,7–9,35,36} It is worth noting that the ORR study for the FeTMP molecule has been done in our previous work.¹⁵ Therefore, we can use our earlier results for the FeTMP substrate to make comparisons, from which we can gauge the new outcomes for the presence of the MoS₂ support on the FeTMP/MoS₂ catalyst.

We designed the MeTMP/MoS₂ substrates (Me = Fe, Co, and Ni). We then optimized the MeTMP/MoS₂ systems with several initial positions of the MeTMP molecule on the optimized MoS₂ surface to search for the most stable configuration of the MeTMP molecule on the MoS₂ support *via* the binding energy calculation:

$$E_b = [E_{\text{Por}} + E_{\text{MoS}_2}] - E_{\text{Por/MoS}_2} \quad (1)$$

where E_{Por} , E_{MoS_2} , and $E_{\text{Por/MoS}_2}$ are the total energies of the isolated MeTMP molecule, the isolated MoS₂ support, and the MeTMP/MoS₂ system, respectively. The configuration with the most positive binding energy is the most thermodynamically stable structure of the MeTMP/MoS₂ system, which shall be selected for exploring the ORR, as presented in the next section.

We employed the Vienna *Ab Initio* Simulation Package (VASP) to perform DFT calculations with the van der Waals corrections (vdW-DF), the revised version of the generalized gradient approximation proposed by Perdew–Burke–Ernzerhof (GGA–revPBE) for the exchange–correlation energy.³⁷ The vdW-DF method was chosen due to its rigorous theoretical foundation, which self-consistently incorporates nonlocal correlation effects.^{38,39} Moreover, this method has demonstrated high accuracy in adsorption energy calculations for similar systems.⁴⁰ The projector-augmented wave technique for pseudopotentials was employed with a plane-wave cutoff energy of 450 eV.⁴¹ The Monkhorst and Pack method was utilized for sampling the special *k*-points with a $3 \times 3 \times 1$ *k*-point mesh grid.⁴² We employed the Gaussian smearing method with a smearing width of 0.1 eV for structural optimization and total energy calculations, and 0.01 eV for calculating the electronic density of states. The criteria for the convergence of the force for atomic position optimization and the energy difference for self-consistent electronic loops are 0.001 eV Å^{−1} and 10^{−5} eV,



respectively. Spin-polarized calculations were performed with a dipole correction applied to the c direction.

The formula used to calculate the adsorption energy of each intermediate (A) on the surface of substrate B is as follows:

$$E_a = E_{AB} - [E_A + E_B]. \quad (2)$$

here, E_{AB} represents the total energy of the optimized substrate-intermediate complex (AB), E_A corresponds to that of the isolated intermediate (A), and E_B denotes the total energy of the clean substrate (B).

The interaction between the substrate and the adsorbed intermediates is often visualized by charge density difference plots,^{43,44} which can be calculated by

$$\Delta\rho = \rho_{ABC} - [\rho_A + \rho_B + \rho_C]. \quad (3)$$

here, ρ_{ABC} , ρ_A , ρ_B , and ρ_C are the charge densities of the full system with support (the intermediate@MeTMP/MoS₂ system), the isolated intermediate, the isolated MeTMP, and the isolated MoS₂ support, respectively. The structures of the A, B, and C components are taken from the optimized ABC system.

The Gibbs free energy (ΔG) for each intermediate step was calculated using the proton and electron exchange model,^{45–48} where the Gibbs free energy involved in the proton and electron combination process is equivalent to that of $\frac{1}{2}\text{H}_2$ in the gas phase at the reversible hydrogen electrode potential, $\text{H}^+ + \text{e}^- = \frac{1}{2}\text{H}_2$:

$$\Delta G(U) = \Delta E + \Delta \text{ZPE} - T\Delta S + neU. \quad (4)$$

where ΔE is the energy difference of the reactant and the product adsorbed on the catalyst surface, which is obtained from our DFT calculations. ΔZPE is the change of the zero-point energies between the reactant and the product in an intermediate step, which is obtained from the vibration frequency calculation for the adsorbed reactant and product. ΔS is the entropy change between the reactant and product in the gas phase, taken from the literature.⁴⁵ n is the number of electrons involved in the intermediate reaction step of the ORR. U is the applied electrode potential referenced to the standard hydrogen electrode. This work was performed for standard conditions, *i.e.*, pH = 0, $p = 1$ bar, and $T = 298$ K.

The Gibbs free energy for the transition state (TS) of the dissociation from O_2^* to 2O^* has been calculated by:

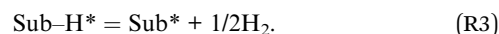
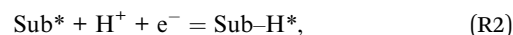
$$\Delta G_{\text{TS}} = \Delta E_{\text{br}} + \Delta \text{ZPE} - T\Delta S. \quad (5)$$

where ΔE_{br} is the energy barrier from O_2^* to (TS). Like the ORR, ΔZPE and ΔS are the change of the zero-point energy and the entropy change between (TS) and O_2^* , respectively. However, the electrode potential does not influence the Gibbs free energy for the transition state because there is no electron reduction in the O_2^* -to- 2O^* dissociation process. The asterisk denotes the adsorbed state of the intermediate on the MeTMP/MoS₂ catalyst.

To consider the possibility that the HER can compete with the ORR on the MeTMP/MoS₂ substrates, we are going to investigate the HER, which is



This reaction on the MeTMP/MoS₂ substrate (Sub) is described as:



For the HER, we calculated the adsorption energy of the H^* intermediate by the equation:

$$E_{\text{ads}} = E_{\text{Sub-H}^*} - \left(E_{\text{Sub}} + \frac{1}{2}E_{\text{H}_2} \right). \quad (6)$$

where $E_{\text{Sub-H}^*}$, E_{Sub} , and E_{H_2} are the total energy for the MeTMP/MoS₂ + H^* system, the isolated MeTMP/MoS₂ substrate, and the hydrogen molecule in the gas phase, respectively. We also calculated the Gibbs free energy for the HER as:

$$\Delta G_{\text{H}^*} = \Delta E_{\text{H}^*} + \Delta \text{ZPE} - T\Delta S + meU. \quad (7)$$

where ΔE_{H^*} is the energy difference between the left and right sides of the reaction (R2). The number of electrons transferred in the HER is m .

Furthermore, we studied the dissolution potential for the Fe, Co, and Ni metals in the MeTMP/MoS₂ systems to estimate the electrochemical stability of the substrates. The potential dissolution reaction is:



The dissolution potential for the metals was calculated by Greeley and Norskov's method,⁴⁹ as follows:

$$U_{\text{diss}} = U_{\text{diss}}^\circ - \Delta E_{\text{f}}/Ze, \quad (8)$$

here, U_{diss}° is the standard dissolution potential of bulk metals from the NIST database,⁵⁰ and Z is the number of electrons released when the metal dissolution occurs. The formation energy ΔE_{f} of the MeTMP/MoS₂ system was calculated by:

$$\Delta E_{\text{f}} = E_{\text{MeTMP/MoS}_2} - (E_{\text{TMP/MoS}_2} + E_{\text{Me}}). \quad (9)$$

where $E_{\text{MeTMP/MoS}_2}$, $E_{\text{TMP/MoS}_2}$, and E_{Me} are the total energy of the MeTMP/MoS₂, TMP/MoS₂ without metal, and the bulk metal, respectively. E_{Me} was obtained from our vdW-DF calculations for the most stable bulk bcc, hcp, and fcc structures of Fe, Co, and Ni, respectively.^{51–53}

3. Results and discussion

3.1 Designing the MeTMP/MoS₂ substrates

Because the metalloporphyrin has a planar structure, we initially set metalloporphyrin at various locations on MoS₂ in the following configurations: (1) inclined, and (2) parallel to the surface of MoS₂. However, we did not consider the vertical configuration of metalloporphyrin because the vacuum space in our system is limited. After optimizing the stacking configurations of MeTMP/MoS₂, we calculated the binding energy of the



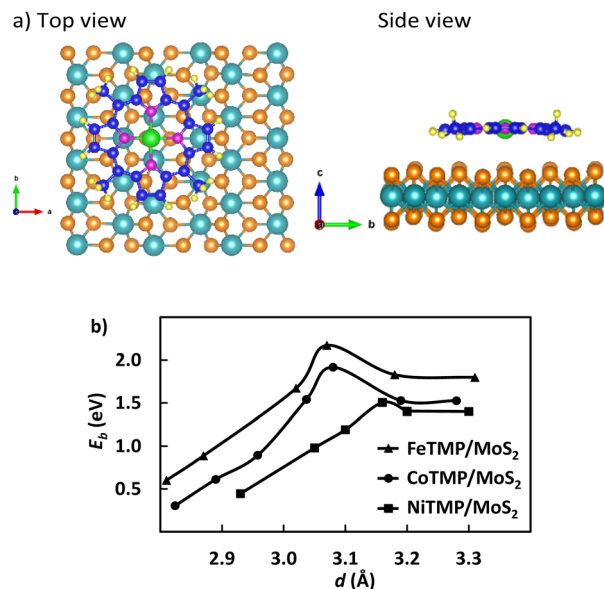


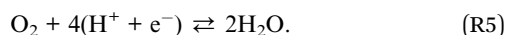
Fig. 1 Structure of the MeTMP/MoS₂ substrate: (a) side view and top view. (b) The binding energy (*E_b*) of the metal porphyrin molecule (MeTMP) on MoS₂ varies with its average distance to the MoS₂ surface, where Me = Fe, Co, and Ni. Light blue (Mo), orange (S), green (Me), dark blue (C), violet (N), and yellow (H).

MeTMP molecule, using expression (1), and its optimized average distance to the MoS₂ surface. We found that the parallel configuration is much more stable than the inclined one for each MeTMP/MoS₂ system.

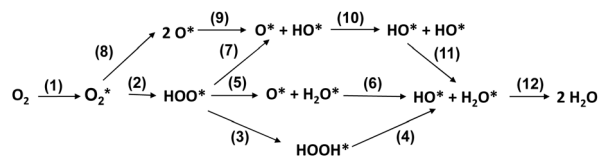
Fig. 1a (top and side views) shows that the porphyrin molecule in the parallel configuration is most stable as its metal atom (green colour) is located on the hollow site of three S atoms in the upper atomic layer of the MoS₂ support. Fig. 1b shows that the binding energy of the porphyrin molecule with the MoS₂ support varies with its average distance (*d*) to the support surface, where the most stable position of the porphyrin molecule corresponds to the highest binding energy *versus* the average distance. We found that the optimized average distance is approximately 3.15, 3.08, and 3.07 Å with the highest binding energy of 1.50, 1.92, and 2.17 eV for the tetra methyl nickel, cobalt, and iron porphyrins, respectively. Moreover, the S atoms on each surface layer are not in the same plane. Several S atoms are shifted upwards and downwards compared to those of the ideal structure (see the side view in Fig. 1a). These most stable MeTMP/MoS₂ systems (Me = Ni, Co, and Fe) were chosen for studying the adsorption of the ORR intermediates and the ORR mechanisms.

3.2 Proposed reaction scenarios and mechanisms

The total ORR is as follows:



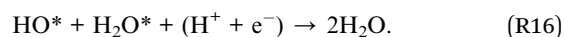
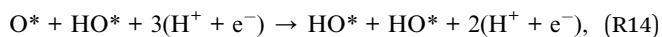
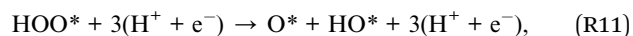
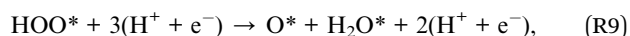
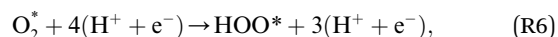
This reaction can proceed through many steps. Scheme 1 shows our proposed possible reaction intermediates and pathways, *i.e.*, step 1: the adsorption of molecular oxygen on the MeTMP/MoS₂ surface transforming O₂ to O₂*. According to the



Scheme 1 Possible pathways for the ORR on the substrates.

hydrogen standard electrode model,⁴⁵ we replaced H⁺ + e[−] by one hydrogen atom (H). Therefore, the proton and electron transfer processes can be modelled *via* the successive hydrogenation steps of the ORR intermediates by sequentially adding a hydrogen atom to an oxygen atom of the previously formed intermediate, starting from transforming O₂* to HOO* (step 2). There are three possibilities for the subsequent step in converting HOO*, *i.e.*, (1) by loading an H atom onto the available O* atom of HOO* to form HOOH* (step 3), and then HO* + H₂O* (step 4); (2) by loading a H atom onto the O atom near the H atom of HOO* to form O* + H₂O* (step 5), and then transform to HO* + H₂O* (step 6); (3) HOO* can also dissociate into O* + HO* (step 7). Besides the molecular oxygen adsorption state, O₂* can also dissociate into two separate oxygen atoms, O* + O* or 2O* (step 8). The hydrogenation of O* + O* can transform it to O* + HO* (step 9), then HO* + HO* (step 10), and HO* + H₂O (step 11). The final product of the ORR is 2H₂O due to transforming HO* + H₂O* (step 12).

From the proposed intermediates and pathways, we can formulate the ORR process explicitly by the following reaction equations:



3.3 Explore the adsorption of ORR intermediates on MeTMP/MoS₂

As shown in Scheme 1, the possible intermediates of the ORR include O₂*, HOO*, HOOH*, O* + H₂O*, HO* + H₂O*, 2O*, O* + HO*, and HO* + HO*. Each reactive intermediate undergoes



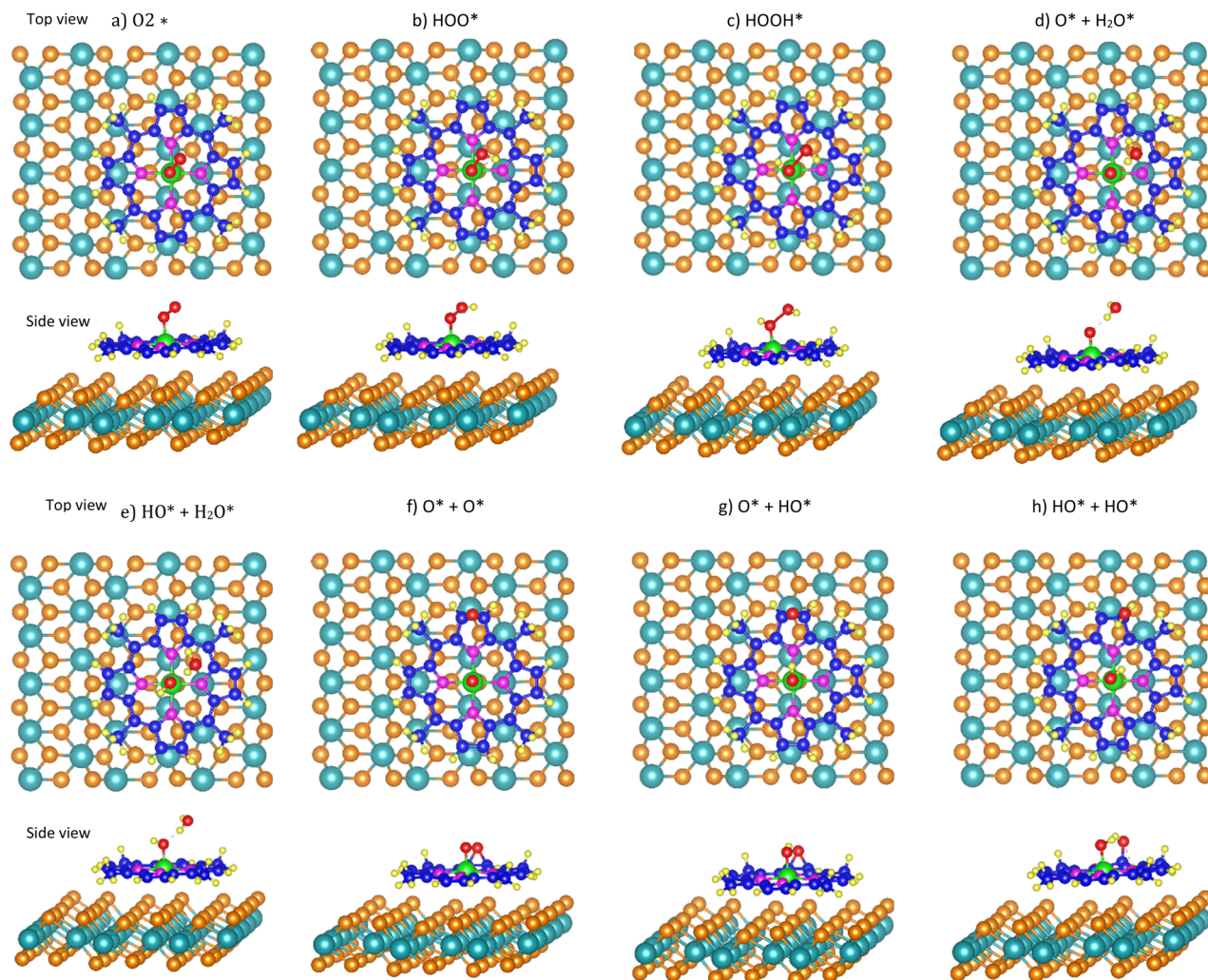


Fig. 2 Most favourable adsorption configuration of the ORR intermediates on the MeTMP/MoS₂ substrates from the top and side views. Light blue (Mo), orange (S), green (Me metal), dark blue (C), violet (N), yellow (H), and red (O).

structural optimization, with the vdW-DF calculations, at various adsorption sites on the MeTMP/MoS₂ surface. The adsorption energy of each configuration is determined *via* formula (2) to assess its adsorption strength. It should be noted that we optimized the structure of O₂, HOO, HOOH, and HO + H₂O to obtain the total energy of these isolated intermediates as a reference for calculating the adsorption energy of O₂*, HOO*, HOOH*, and HO* + H₂O*, respectively. However, the isolated O + O, O + HO, and HO + HO (O + H₂O) intermediates do not exist naturally in the gas phase, so they have no physical meaning to use as references. Therefore, we used the total energy of the isolated O₂, HOO, and H₂O₂ for calculating the adsorption energy of the O* + O*, O* + HO*, and HO* + HO* (O* + H₂O*) intermediates, respectively.

According to the definition of eqn (2), the more negative the adsorption energy, the more stable the adsorption is. We found that the most favourable adsorption site for the ORR intermediates is around the metal site (Me) of the MeTMP molecules of the cobalt, nickel, and iron tetramethyl porphyrins/MoS₂ substrates (Fig. 2). We found that the molecular-oxygen-

containing intermediates, *i.e.*, O₂*, HOO*, and HOOH* adsorbed with the end-on configurations, where one oxygen atom is on the top of the metal atom Me and another one is over the hollow site of the Me–N–C ring of the MeTMP molecules (Fig. 2a–c). Meanwhile, the atomic-oxygen-containing intermediates such as O* + H₂O*, HO* + H₂O*, 2O*, O* + HO*, and HO* + HO* adsorbed most stably with one oxygen atom always on the top of the metal atom (Me), and the other oxygen atom can be over the hollow site or another place of the porphyrin ring (Fig. 2d–h).

Table S1† shows the bond distances from the nearest oxygen atom to the metal atom Me of the MeTMP molecule, $d_{\text{O-M}}$, and between two oxygen atoms of the intermediates, $d_{\text{O-O}}$. We also found the values of these bond distances for the FeTMP substrate in our previous publication.¹⁵ For all MeTMP/MoS₂ substrates, we found that the $d_{\text{O-M}}$ value of O₂* (row 1) is always significantly larger than that of 2O* (row 6). Meanwhile, $d_{\text{O-M}}$ of HOO* (row 2, column 2) > $d_{\text{O-M}}$ of O* + HO* (row 7, column 2) and $d_{\text{O-M}}$ of HOOH* (row 3, column 2) > $d_{\text{O-M}}$ of 2HO* (row 8, column 2) for the NiTMP/MoS₂ substrate, and $d_{\text{O-M}}$ of HOO* ≈



Table 1 Adsorption energy (eV) of the ORR intermediates on metal tetra-methyl porphyrins (MeTMP, where Me = Ni, Co, and Fe) with and without MoS₂ support

Intermediates	NiTMP/MoS ₂	CoTMP/MoS ₂	FeTMP/MoS ₂	NiTMP	CoTMP
O ₂ *	−0.15	−0.68	−0.48	−0.20	−0.75
HOO*	−0.61	−1.29	−1.30	−0.44	−1.29
HOOH*	−0.05	−0.37	−0.44	−0.29	−0.46
O* + H ₂ O*	0.13	−1.48	−2.06	−0.06	−1.40
HO* + H ₂ O*	−1.84	−2.61	−2.79	−1.73	−2.65
2O*	1.14	−0.41	−0.96	1.03	−0.01
O* + HO*	−1.13	−1.99	−2.13	−0.92	−1.89
2HO*	−0.38	−1.44	−1.45	−0.19	−1.13

d_{O-M} of O* + HO* and d_{O-M} of HOOH* \approx d_{O-M} of 2HO* for the CoTMP/MoS₂ and FeTMP/MoS₂ substrates. Remarkably, the d_{O-O} value of the molecular-oxygen-containing intermediates is always much shorter than that of the atomic-oxygen-containing intermediates with the same number of hydrogen atoms, *i.e.*, d_{O-O} of O₂* \ll d_{O-O} of 2O*, d_{O-O} of HOO* \ll d_{O-O} of O* + HO*, and d_{O-O} of HOOH* \ll d_{O-O} of 2HO* for all the MeTMP/MoS₂ substrates. It should be noted that the O* + H₂O* and HO* + H₂O* intermediates belong to both molecular and atomic oxygen adsorption scenarios, which have the d_{O-M} and d_{O-O} bond distances in the middle range between that of the molecular-oxygen-containing intermediates and that of the atomic-oxygen-containing ones for all the MeTMP/MoS₂ substrates.

Table 1 presents the adsorption energy of the ORR intermediates in their most favourable configuration on the substrates. The negative and positive adsorption energies indicate the thermodynamically favourable and unfavourable adsorption of the intermediates, respectively. Therefore, we found in Table 1 that the O* + H₂O* and 2O* intermediates on NiTMP/MoS₂ and 2O* on NiTMP are unfavourable. For each substrate with and without the MoS₂ support, HO* + H₂O* has the greatest negative adsorption energy among all the ORR intermediates. We also compare the adsorption energy calculated by the same vdW-DF scheme for the FeTMP molecule in ref. 15. Notably, the substitution of Ni, Co, and Fe metals into the tetramethyl porphyrin molecule causes a modification in the adsorption energy in increasingly negative order: Ni < Co < Fe for all the intermediates except for O₂*, which follows the order: Ni < Fe < Co on all the MeTMP/MoS₂ and MeTMP substrates. This order has a similar trend as the previous study,⁵⁴ where the authors investigated the ORR on phenyl metalloporphyrins and metal phthalocyanines without supports.

In the presence of the MoS₂ support, the adsorption energy of all the ORR intermediates on FeTMP/MoS₂ uniquely becomes less negative than that on the FeTMP (ref. 15). However, the MoS₂ support affects the adsorption of the intermediates differently for the Ni and Co substitutions. By comparing the second with the fifth and third with the sixth columns of Table 1, we found that the adsorption strength of O₂*, HOOH*, O* + H₂O*, and 2O* on NiTMP/MoS₂, O₂*, HOO*, HOOH*, and HO* + H₂O* on CoTMP/MoS₂ decreases and otherwise increases, compared to that on NiTMP and CoTMP, respectively.

3.4 Electronic properties

Analysing the electronic structural characteristics, such as the Bader charge and the charge density difference, can expose the physical insights into the interaction between the ORR intermediates and the MeTMP/MoS₂ substrates. The Bader charge in Table S2† indicates that the metal Me and C atoms of the MeTMP molecule consistently donate the negative charge (e[−]). Meanwhile, the N and H atoms of the MeTMP molecule always accumulate the charge for every case, including having the adsorption of the ORR intermediates. However, in the presence of the MoS₂ support, the H atoms of the MeTMP molecule can donate or accumulate the charge for different intermediates. As a result, the clean MeTMP molecules (without MoS₂ support) retain their neutral charge (0.000 e[−]). However, they become positively charged in the presence of the MoS₂ support or they donate the negative charge of about 0.5 e[−] to the MoS₂ support, thereby strengthening the bond between the MeTMP molecule and MoS₂. The charge gain of MoS₂ can be arranged in the following order of magnitude: NiTMP/MoS₂ < CoTMP/MoS₂ < FeTMP/MoS₂, which is consistent with the trend of the maximum binding energy of the MeTMP molecule on the MoS₂ support, as analysed in Section 3.1. In the presence of the ORR intermediates, the MeTMP molecules with and without the MoS₂ support always donate the charge to the intermediates for all cases. Moreover, the MeTMP molecule always transfers a significant amount of its charge to the MoS₂ support.

For each case of intermediate adsorption, by comparing MeTMP/MoS₂ to MeTMP (the Bader charge for the FeTMP substrate found in Table 2, column 4 of ref. 15), we found that the MoS₂ support causes the Me and C atoms to donate more and less charge, except for Me=Fe in the HOOH*@FeTMP/MoS₂ system and C atoms in the [O* + HO*]@NiTMP/MoS₂ system, respectively. The MoS₂ support causes the N and H atoms to gain less charge for every case of the ORR intermediate adsorption when compared to the MeTMP molecule without the MoS₂ support. Furthermore, the charge accumulation of the dissociative intermediates is significantly higher than that of the associative intermediates, *i.e.*, 2O* > O₂*, O* + HO* > HOO*, and HO* + HO* > HOOH*. The charge gains of O* + H₂O* and HO* + H₂O* lie within the middle range of that of the dissociative and associative intermediates. This agrees with the trend of their d_{O-M} and d_{O-O} bond distances, as analysed previously. Upon adding the MoS₂ support, we observed a reduction in the charge accumulation of the ORR intermediates, particularly in



Table 2 Activation barrier (eV) for the ORR steps at $U = 1.23$ V, i.e., the Gibbs free energy difference of the unfavourable step compared to the previous one, and $\frac{1}{2}\text{H}_2 = \text{H}^+ + \text{e}^-$

ORR intermediates	NiTMP/MoS ₂	CoTMP/MoS ₂	FeTMP/MoS ₂	NiTMP	CoTMP
$\text{O}_2^* + 2\text{H}_2 \rightarrow \text{HOO}^* + 3/2\text{H}_2$	1.20	1.05	0.84	1.15	1.10
$\text{HOO}^* + 3/2\text{H}_2 \rightarrow \text{HOOH}^* + \text{H}_2$	0.59	1.08	1.03	0.60	1.01
$\text{HOO}^* + 3/2\text{H}_2 \rightarrow \text{O}^* + \text{H}_2\text{O}^* + \text{H}_2$	0.99	0.06	0.00	0.89	0.15
$\text{O}^* + \text{H}_2\text{O}^* + \text{H}_2 \rightarrow \text{HO}^* + \text{H}_2\text{O}^* + \frac{1}{2}\text{H}_2$	0.00	0.06	0.46	0.00	0.00
$\text{HO}^* + \text{H}_2\text{O}^* + \frac{1}{2}\text{H}_2 \rightarrow 2\text{H}_2\text{O}$	0.00	0.00	0.01	0.00	0.00
$\text{O}_2^* + 2\text{H}_2 \rightarrow 2\text{O}^* + 2\text{H}_2$	3.45	0.92	1.82	4.85	3.34
$2\text{O}^* + 2\text{H}_2 \rightarrow \text{O}^* + \text{HO}^* + 3/2\text{H}_2$	0.00	0.07	0.49	0.95	0.92
$\text{O}^* + \text{HO}^* + 3/2\text{H}_2 \rightarrow \text{HO}^* + \text{HO}^* + \text{H}_2$	0.00	0.00	0.85	0.00	0.00
$\text{O}^* + \text{HO}^* + 3/2\text{H}_2 \rightarrow \text{O}^* + \text{H}_2\text{O}^* + \text{H}_2$	1.48	0.74	0.29	0.10	0.00

the case of $\text{O}^* + \text{HO}^*$ and $\text{HO}^* + \text{HO}^*$. Most importantly, we found that the charge exchange between the MeTMP and MeTMP/MoS₂ substrates with the ORR intermediates is significant. Therefore, we can conclude that the nature of the interaction between the intermediates and the substrates is due to the charge exchange.

The charge density difference of the MeTMP/MoS₂ substrates with the adsorbed ORR intermediates, calculated by the formula (3) (Fig. S2–S4 in ESI†), shows that the centre of the charge gain is always the oxygen atoms of the ORR intermediates. While the metal and C atoms of MeTMP porphyrins donate, the upper atomic layer of the S atoms gains the charge. These results agree with the Bader charge analysis, as shown above.

3.5 Mechanism of oxygen reduction reaction on MeTMP/MoS₂

The Gibbs free energy for each side of the reactions (R6) to (R16) was calculated relative to that of two water molecules, which are the final product of the ORR. The Gibbs free energy was studied at the standard condition. The values are listed in Table S3.† Dissociating O_2^* into 2O^* is possible if overcoming an energy barrier at the transition state (TS), E_{TS} , which was identified using the nudged elastic bands method, as shown in Fig. 3 for MeTMP/MoS₂ and Fig. S5† for MeTMP. The calculation involved the initial and final states (denoted as (IS) and (FS)) corresponding to the most stable molecular adsorption configuration of the O_2^* molecule and the most stable atomic adsorption

configuration of 2O^* . Once the transition state structure was determined, the total energy and the Gibbs free energy of (TS) + 2H_2 were calculated following eqn (5).

We can write the reaction series as, $\text{O}_2^* + 2\text{H}_2 \rightarrow (\text{TS}) + 2\text{H}_2 \rightarrow 2\text{O}^* + 2\text{H}_2$. The energy barrier for the dissociation process of O_2^* into 2O^* , which is the energy of the (TS) state relative to that of the (IS) state, is 3.69, 1.43, 1.94, 5.89, and 4.79 eV on NiTMP/MoS₂, CoTMP/MoS₂, FeTMP/MoS₂, NiTMP, and CoTMP, respectively. The energy barrier of 2.67 eV for this process on FeTMP was given in our previous publication.¹⁵ We found that the MoS₂ support could significantly reduce the energy barrier for the O_2^* dissociation. Therefore, it can facilitate the dissociative pathway of the ORR.

Fig. 4 displays the Gibbs free energy diagrams of ORR for the associative (left panel) and the dissociative (right panel) mechanisms on MeTMP/MoS₂ (upper panel) and MeTMP (lower panel). As shown in Fig. 4a–d, forming $\text{O}_2^* + 2\text{H}_2$ on all of the substrates is automatic and does not require any activation energy. Fig. 4a shows that the first two hydrogenation steps go uphill, except for the second hydrogenation converting $\text{HOO}^* + 3/2\text{H}_2$ to $\text{O}^* + \text{H}_2\text{O}^* + \text{H}_2$ (the red dashed line) on the FeTMP/MoS₂ substrate, indicating the need for an activation energy. The third hydrogenation step converting $\text{HOOH}^* + \text{H}_2$ and $\text{O}^* + \text{H}_2\text{O}^* + \text{H}_2$ to $\text{HO}^* + \text{H}_2\text{O}^* + \frac{1}{2}\text{H}_2$ goes downhill, except for transforming $\text{O}^* + \text{H}_2\text{O}^* + \text{H}_2$ to $\text{HO}^* + \text{H}_2\text{O}^* + \frac{1}{2}\text{H}_2$ on the CoTMP/MoS₂ and FeTMP/MoS₂ substrates. The fourth hydrogenation goes downhill on NiTMP/MoS₂ and CoTMP/MoS₂, and

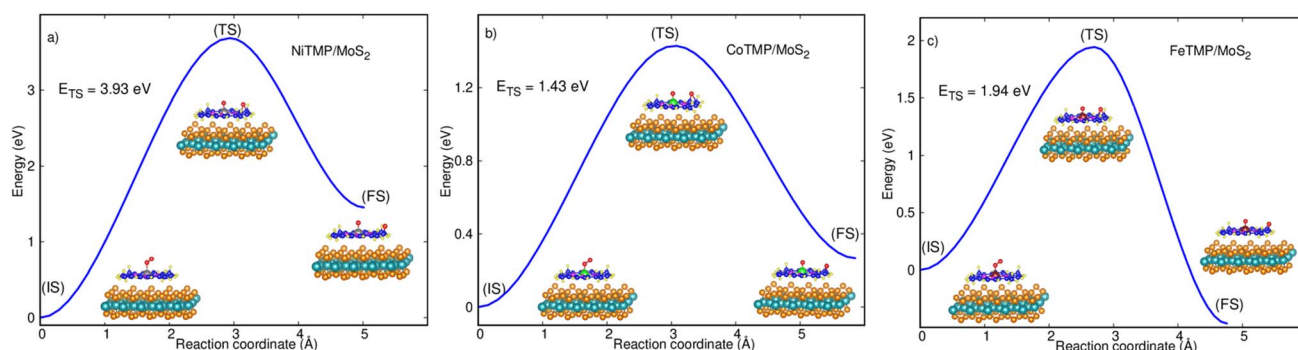


Fig. 3 Dissociating O_2^* to 2O^* , from the initial state (IS) overcoming the transition state (TS) to the final state (FS), on NiTMP/MoS₂ (a), CoTMP/MoS₂ (b), and FeTMP/MoS₂ (c). Light blue (Mo), orange (S), gray (Ni), green (Co), brown (Fe), dark blue (C), violet (N), yellow (H), and red (O).



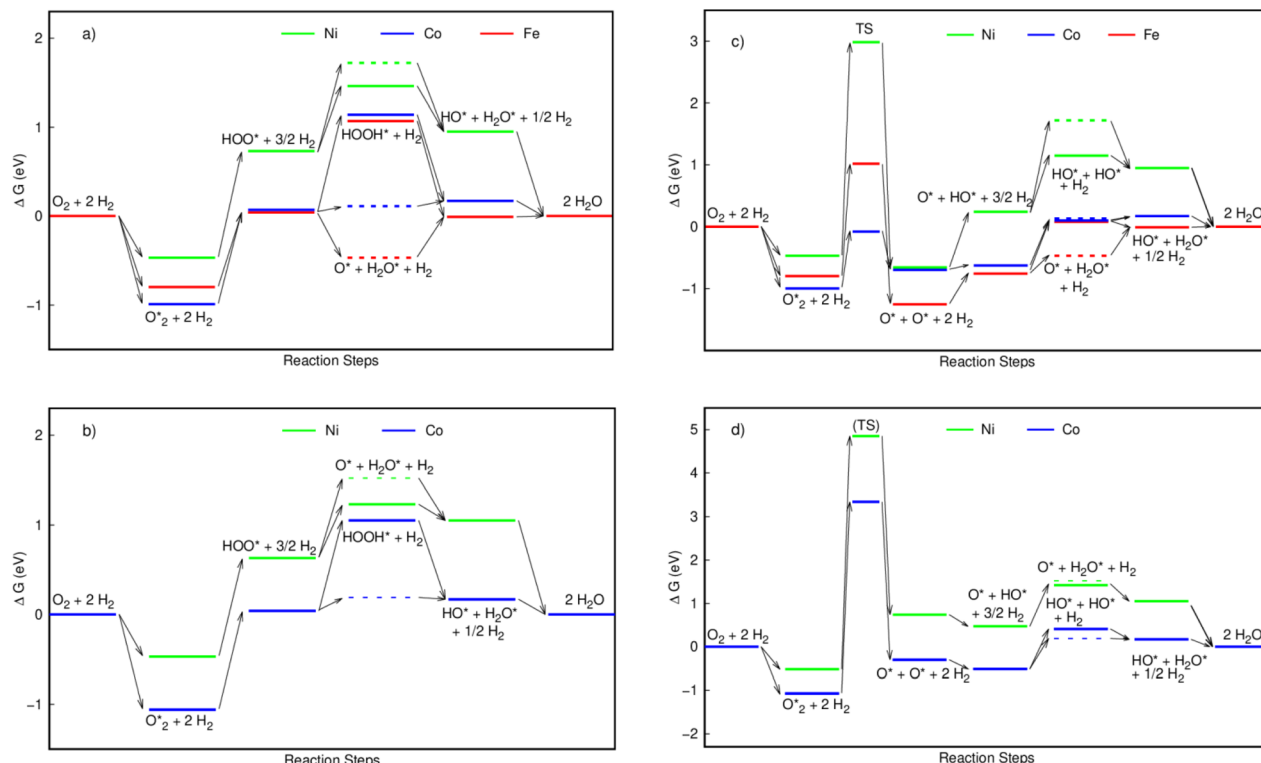


Fig. 4 Relative Gibbs energy diagrams for the associative pathways on MeTMP/MoS₂ (a) and MeTMP (b), and the dissociative pathways on MeTMP/MoS₂ (c) and MeTMP (d) at the reversible electrode potential of $U = 1.23$ V. The dashed lines depict the Gibbs free energy level of $O^* + H_2O^* + H_2$, while the solid lines depict the remaining ones.

slightly uphill on FeTMP/MoS₂. We also found the same characteristics for the Gibbs free energy diagrams of the associative pathway on MeTMP without MoS₂ support (Fig. 4b) with a slight increase of the energy level of $O^* + H_2O^* + H_2$ on CoTMP, so that it is almost the same height as that of $HO^* + H_2O^* + \frac{1}{2}H_2$ on CoTMP. The downhill steps do not require activation energy, and can proceed automatically.

For dissociation mechanisms on MeTMP/MoS₂ (Fig. 4c), to proceed in the forward direction (from left to right of the diagram), the ORR must overcome an activation barrier at the dissociation step converting O_2^* to $2O^*$. Afterward, the first, second, third, and fourth hydrogenation steps go uphill except for the third hydrogenation that converts $HO^* + HO^* + H_2$ and $O^* + H_2O^* + H_2$ to $HO^* + H_2O^* + \frac{1}{2}H_2$ on NiTMP/MoS₂, and the fourth hydrogenation that transforms $HO^* + H_2O^* + \frac{1}{2}H_2$ to $2H_2O$ on the NiTMP/MoS₂ and CoTMP/MoS₂ substrate. For the dissociation mechanisms on MeTMP without MoS₂ support (Fig. 4d), the relative position of the Gibbs free energy levels is more complicated. However, the most striking feature is that the activation barrier for the O_2^* -to- $2O^*$ dissociation is much higher than that on MeTMP with the MoS₂ support.

For the backward direction (from right to left of each diagram), the Gibbs free energy levels of $HO^* + H_2O^* + \frac{1}{2}H_2$ and $O^* + H_2O^* + H_2$ in the associative mechanisms (Fig. 4a and b) and $HO^* + H_2O^* + \frac{1}{2}H_2$, $O^* + H_2O^* + H_2$, $O^* + HO^* + \frac{3}{2}H_2$, and $O^* + O^* + 2H_2$ in the dissociative mechanisms on the FeTMP/MoS₂ substrates are lower than that of $2H_2O$. These results

indicate that the backward processes of the ORR may occur, and therefore hinder the catalytic efficiency of the FeTMP/MoS₂ substrate. The activation barrier for the associative and dissociative mechanisms on various substrates is listed in Table 2. The highest activation barrier determines the rate-limiting step for each reaction pathway. For the associative pathways, we found that the rate-limiting step occurs at the first hydrogenation step for NiTMP/MoS₂ and NiTMP, and the second hydrogenation step for FeTMP/MoS₂. However, the presence of the MoS₂ support modified the rate-limiting step of the ORR from the first hydrogenation on CoTMP to the second hydrogenation on CoTMP/MoS₂.

For the dissociative mechanisms, we found that the rate-limiting step always occurs at the dissociating step, forming $2O^* + 2H_2$ from $O_2^* + 2H_2$ for all substrates. The highest thermodynamic barrier for the dissociative mechanisms on NiTMP/MoS₂, CoTMP/MoS₂, and FeTMP/MoS₂ is 3.45, 0.92, and 1.82 eV, which is much lower than 4.85, 3.34, and 2.19 eV on NiTMP,

Table 3 Adsorption energy (eV) of H^* on the MeTMP/MoS₂ substrate. The optimized structures of the H^* adsorption configuration on the MeTMP/MoS₂ substrates are shown in Fig. S6

Adsorption site	CoTMP/MoS ₂	FeTMP/MoS ₂	NiTMP/MoS ₂
Metal (Fe, Co, Ni) atom	0.17	0.54	0.89
S atom	0.12	0.12	0.17
Mo atom	1.05	1.03	1.03
S-S bridge	0.68	0.57	0.78



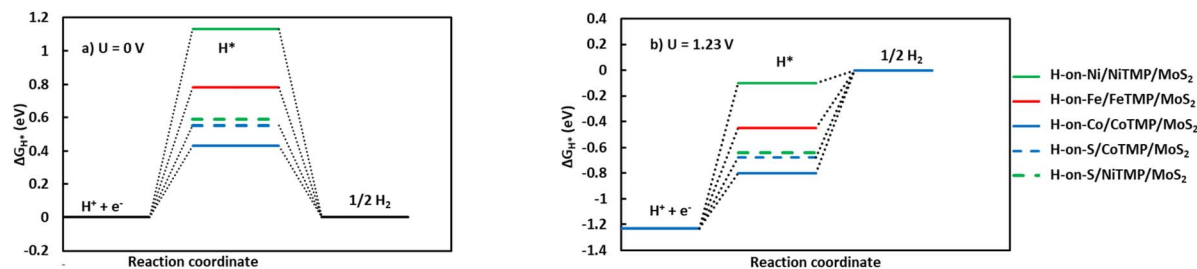


Fig. 5 Gibbs free diagram of the HER for the most favourable configurations of H* on MeTMP/MoS₂ at zero potential (a) and the equilibrium potential (b). The line for H-on-S/FeTMP/MoS₂ is identical to that for H-on-S/CoTMP/MoS₂.

CoTMP, and FeTMP,¹⁵ respectively. We also found that the MoS₂ support does not lower the energy barrier for the associative mechanisms. However, it does significantly lower the energy barrier for the dissociative ones. In particular, it made the CoTMP/MoS₂ and FeTMP/MoS₂ systems the best and second-best catalysts among the considered ones for the dissociative mechanisms, respectively. Notably, nickel does not favour the ORR for both mechanisms. Considering all forward and backward directions, we can conclude that CoTMP/MoS₂ is the superior catalyst.

The superior catalytic activity of CoTMP/MoS₂ is because of its most significant reduction in the thermodynamic energy barrier, as seen in Fig. 4c, which stems from its lowest energy level of the transition state for the O₂* to 2O* conversion. Also, the electronic configuration of the Fe, Co, and Ni atoms is [Ar] 3d⁶4s², [Ar] 3d⁷4s², and [Ar] 3d⁸4s², respectively. We find that the number of electrons existing in the 3d orbital of the valence state of the Fe, Co, and Ni atoms is 6, 7, and 8, respectively. We guess that the odd number of electrons in the 3d orbital of the Co atom is responsible for the superior catalytic activity of CoTMP/MoS₂ over the FeTMP/MoS₂ and NiTMP/MoS₂ substrates.

For the HER, we optimized the position of H* at all possible sites, such as on the metal (Fe, Co, and Ni), S atom, Mo atom, S-S bridge, and hollows. We found the optimized configuration, as shown in Fig. S6.† The positive adsorption energy of H* (Table 3) indicates that the adsorption of H* is unfavourable on the MeTMP/MoS₂ substrates, which differs from the negative adsorption energy of the ORR intermediates. We then calculated the Gibbs free energy of reaction steps (R2) and (R3) under standard thermodynamic conditions (pH = 0, 298 K, and 1.0 bar). According to reactions (R1) and (R2), we have $m = 1$ and $\Delta E_{H^*} = E_{ads}$.

Fig. 5 shows that HER is thermodynamically unfavourable when it requires an activation energy. This energy is significantly lower than or comparable to the activation barrier of about 1.0 eV for the ORR at the electrode potential of $U = 1.23$ V. However, the reverse process of HER occurs spontaneously for all of the

substrates. Therefore, there are always protons on the surface of the substrates, indicating that the HER does not interfere with ORR.

The dissolution potential, presented in Table 4, shows that the Co, Fe, and Ni metals are stabilized in the MeTMP/MoS₂ system because their dissolution potential is much higher than the equilibrium potential of 1.23 V for the ORR.

4. Conclusions

Using vdW-DF with the revised PBE version of the generalized gradient approximation and thermodynamic model, this study elucidated the impact of metal substitution in MeTMP/MoS₂ on the overall catalytic performance of the ORR. The Co atom is the best substitution in the MeTMP/MoS₂ catalysts. In the association mechanisms, the first and second hydrogenation steps emerge as the rate-determining steps for NiTMP/MoS₂ (NiTMP) and FeTMP/MoS₂, respectively. However, the rate-limiting step is the first hydrogenation for CoTMP and the second hydrogenation for CoTMP/MoS₂. Notably, the dissociation mechanisms have the rate-determining step occurring at the O₂*-to-2O* dissociation on all substrates. The metals in the MeTMP/MoS₂ catalyst are stable due to their high dissolution potentials. Moreover, the HER does not interfere with the ORR. The charge exchange dominates the interaction between the ORR intermediates and the substrates.

Data availability

The data supporting this article have been included in the ESI.†

Author contributions

Conceptualization (DNS), investigation (TPD), formal analysis (DNS), resources (DNS and VC), supervision (DNS, PTNN, and VC), validation (DNS), visualization (TPD), writing manuscript (DNS and TPD), and reviewing and editing (DNS, TPD, PTNN and VC).

Conflicts of interest

There are no conflicts of interest to declare.

Table 4 Formation energy and dissolution potential of MeTMP/MoS₂

Metal	ΔE_f (eV)	E_{Me} (eV)	U_{diss}^{50} (V)	Z^{54}	U_{diss} (V)
Co	−4.97	−3.62	−0.28	2	2.21
Fe	−4.55	−4.92	−0.45	2	1.82
Ni	−5.16	−1.91	−0.26	2	2.32



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References

- 1 H. A. Gasteiger, S. S. Kocha, B. Sompalli and F. T. Wagner, Activity benchmarks and requirements for Pt, Pt-alloy, and non-Pt oxygen reduction catalysts for PEMFCs, *Appl. Catal., B*, 2005, **56**, 9–35.
- 2 F. Cheng and J. Chen, Metal-air batteries: from oxygen reduction electrochemistry to cathode catalysts, *Chem. Soc. Rev.*, 2012, **41**, 2172–2192.
- 3 L. Jiang, M. Li, L. Lin, Y. Li, X. He and L. Cui, Electrocatalytic activity of metalloporphyrins grown in situ on graphene sheets toward oxygen reduction reaction in an alkaline medium, *RSC Adv.*, 2014, **4**, 26653–26661.
- 4 D. N. Son, H. Nakanishi, M. Y. David and H. Kasai, Oxygen Reduction on Pt(111) Cathode of Fuel Cells, *J. Phys. Soc. Jpn.*, 2009, **78**, 114601.
- 5 D. N. Son, O. K. Le, V. Chihaia and K. Takahashi, Effects of Co Content in Pd-Skin/PdCo Alloys for Oxygen Reduction Reaction: Density Functional Theory Predictions, *J. Phys. Chem. C*, 2015, **119**, 24364–24372.
- 6 D. N. Son, P. N. Thanh, N. D. Quang, K. Takahashi and M. P. Pham-Ho, First-principles study of Pd-skin/Pd₃Fe(111) electrocatalyst for oxygen reduction reaction, *J. Appl. Electrochem.*, 2017, **47**, 747–754.
- 7 C. J. Kaminsky, J. Wright and Y. Surendranath, Graphite-conjugation enhances porphyrin electrocatalysis, *ACS Catal.*, 2019, **9**, 3667–3671.
- 8 A. N. Marianov, A. S. Kochubei, S. Gu and Y. Jiang, Charge-Transfer Mechanism in Oxygen Reduction over Co Porphyrins: Single-Site Molecular Electrocatalysts to Macromolecular Frameworks, *ACS Catal.*, 2022, **12**, 8610–8622.
- 9 Y. Liu, G. Zhou, Z. Zhang, H. Lei, Z. Yao, J. Li, J. Lin and R. Cao, Significantly improved electrocatalytic oxygen reduction by an asymmetrical Pacman dinuclear cobalt (II) porphyrin–porphyrin dyad, *Chem. Sci.*, 2020, **11**, 87–96.
- 10 G. Passard, D. K. Dogutan, M. Qiu, C. Costentin, *et al.*, Oxygen Reduction Reaction Promoted by Manganese Porphyrins, *ACS Catal.*, 2018, 8671–8679.
- 11 D.-H. Kim, D.-H. Kwak, S.-B. Han, *et al.*, Electrochemical catalytic contribution of transition metals at the center of porphyrin macrocycle structures as catalysts for oxygen reduction reaction, *J. Ind. Eng. Chem.*, 2017, **54**, 200–204.
- 12 L. Ye, Y. Fang, Z. Ou, S. Xue and K.-M. Kadish, Cobalt tetrabutano- and tetrabenzotetraarylporphyrin complexes: effect of substituents on the electrochemical properties and catalytic activity of oxygen reduction reactions, *Inorg. Chem.*, 2017, **56**, 13613–13626.
- 13 B. Sun, Z. Ou, D. Meng, *et al.*, Electrochemistry and catalytic properties for dioxygen reduction using ferrocene-substituted cobalt porphyrins, *Inorg. Chem.*, 2014, **53**, 8600–8609.
- 14 S. Amanullah, P. Saha and A. Dey, O₂ reduction by iron porphyrins with electron withdrawing groups: to scale or not to scale, *Faraday Discuss.*, 2022, **234**, 143–158.
- 15 T. P. Dung, V. Chihaia and D. N. Son, Effects of functional groups in iron porphyrin on the mechanism and activity of oxygen reduction reaction, *RSC Adv.*, 2023, **13**, 8523–8534.
- 16 H. Wan, T. M. Østergaard, L. Arnarson and J. Rossmeisl, Climbing the 3D volcano for the oxygen reduction reaction using porphyrin motifs, *ACS Sustainable Chem. Eng.*, 2018, **7**, 611–617.
- 17 G. Tei, T. Tamaki, T. Hayashi, *et al.*, Oxygen Reduction Reaction (ORR) Activity of a Phenol-Substituted Linear FeIII-Porphyrin Dimer, *Eur. J. Inorg. Chem.*, 2017, **26**, 3229–3232.
- 18 T. Schilling, A. Okunola, J. Masa, W. Schuhmann and M. Bron, Carbon nanotubes modified with electrodeposited metal porphyrins and phenanthrolines for electrocatalytic applications, *Electrochim. Acta*, 2010, **55**, 7597–7602.
- 19 M. Jahan, Q. Bao and K. P. Loh, Electrocatalytically Active Graphene–Porphyrin MOF Composite for Oxygen Reduction Reaction, *J. Am. Chem. Soc.*, 2012, **134**, 6707–6713.
- 20 L. Wei and Y. Chen, Degradation of carbon materials in electrocatalysis, *Curr. Opin. Electrochem.*, 2022, **36**, 101159.
- 21 J. Kwon, P. Choi, S. Jo, H. Oh, K. Y. Cho, Y. K. Lee, *et al.*, Identification of electrode degradation by carbon corrosion in polymer electrolyte membrane fuel cells using the distribution of relaxation time analysis, *Electrochim. Acta*, 2022, **414**, 140219.
- 22 S. Jayabal, G. Saranya, J. Wu, Y. Liu, *et al.*, Understanding the high-electrocatalytic performance of two-dimensional MoS₂ nanosheets and their composite materials, *J. Mater. Chem. A*, 2017, **5**, 24540–24563.
- 23 I. S. Amiin, Z. Pu, X. Liu, *et al.*, Multifunctional Mo-N/C@MoS₂ Electrocatalysts for HER, OER, ORR, and Zn-Air Batteries, *Adv. Funct. Mater.*, 2017, **27**, 1702300.
- 24 Q. Hua, K. E. Madsen, A. M. Esposito, *et al.*, Effect of support on oxygen reduction reaction activity of supported iron porphyrins, *ACS Catal.*, 2022, **12**, 1139–1149.
- 25 C. Suresh, S. Mutyala and J. Mathiyarasu, Support interactive synthesis of nanostructured MoS₂ electrocatalyst for oxygen reduction reaction, *Mater. Lett.*, 2016, **164**, 417–420.
- 26 M. T. Anwar, X. Yan, M. R. Asghar, *et al.*, MoS₂-rGO hybrid architecture as durable support for cathode catalyst in proton exchange membrane fuel cells, *Chin. J. Catal.*, 2019, **40**, 1160–1167.
- 27 D. Sun, D. Huang, H. Wang, *et al.*, 1T MoS₂ nanosheets with extraordinary sodium storage properties via thermal-driven ion intercalation assisted exfoliation of bulky MoS₂, *Nano Energy*, 2019, **61**, 361–369.
- 28 D. Hou, W. Zhou, X. Liu, *et al.*, Pt nanoparticles/MoS₂ nanosheets/carbon fibers as efficient catalyst for the hydrogen evolution reaction, *Electrochim. Acta*, 2015, **166**, 26–31.
- 29 I. S. Kwon, I. H. Kwak, J. Kim, *et al.*, Two-Dimensional MoS₂/Fe-Phthalocyanine Hybrid Nanostructures as Excellent



- Electrocatalysts for Hydrogen Evolution and Oxygen Reduction Reactions, *Nanoscale*, 2019, **11**, 14266–14275.
- 30 L. Hao, J. Yu, X. Xu, *et al.*, Nitrogen-doped MoS₂/carbon as highly oxygen-permeable and stable catalysts for oxygen reduction reaction in microbial fuel cells, *J. Power Sources*, 2017, **339**, 68–79.
 - 31 G. Gao, Y. Jiao, F. Ma, *et al.*, Charge Mediated Semiconducting-to-Metallic Phase Transition in Molybdenum Disulfide Monolayer and Hydrogen Evolution Reaction in New 1T' Phase, *J. Phys. Chem. C*, 2015, **119**, 13124–13128.
 - 32 Y. Yu, *et al.*, High phase-purity 1T'-MoS₂- and 1T'-MoSe₂-layered crystals, *Nat. Chem.*, 2018, **10**, 638–643.
 - 33 A. Ejigu, I. A. Kinloch, E. Prestat, *et al.*, A simple electrochemical route to metallic phase trilayer MoS₂: evaluation as electrocatalysts and supercapacitors, *J. Mater. Chem. A*, 2017, **5**, 11316–11330.
 - 34 X. Qian, J. Liu, L. Fu and J. Li, Quantum spin Hall effect in two-dimensional transition metal dichalcogenides, *Science*, 2014, **346**, 1344–1347.
 - 35 M. S. Liao and S. Scheiner, Electronic structure and bonding in metal porphyrins, metal=Fe, Co, Ni, Cu, Zn, *J. Chem. Phys.*, 2002, **117**(1), 205–219.
 - 36 K. Sengupta, S. Chatterjee and A. Dey, In situ mechanistic investigation of O₂ reduction by iron porphyrin electrocatalysts using surface-enhanced resonance Raman spectroscopy coupled to rotating disk electrode (SERRS-RDE) setup, *ACS Catal.*, 2016, **6**(10), 6838–6852.
 - 37 J. Hafner, Ab-initio simulations of materials using VASP: Density-functional theory and beyond, *J. Comput. Chem.*, 2008, **29**, 2044–2078.
 - 38 M. Dion, H. Rydberg, E. Schröder, D. C. Langreth and B. I. Lundqvist, Van der Waals Density Functional for General Geometries, *Phys. Rev. Lett.*, 2004, **92**, 246401.
 - 39 J. Hermann, R. A. DiStasio and A. Tkatchenko, First-Principles Models for van der Waals Interactions in Molecules and Materials: Concepts, Theory, and Applications, *Chem. Rev.*, 2017, **117**, 4714–4758.
 - 40 P. G. Moses, J. J. Mortensen, B. I. Lundqvist and J. K. Nørskov, Density functional study of the adsorption and van der Waals binding of aromatic and conjugated compounds on the basal plane of MoS₂, *J. Chem. Phys.*, 2009, **130**, 104709.
 - 41 P. E. Blöchl, Projector augmented-wave method, *Phys. Rev. B:Condens. Matter Mater. Phys.*, 1994, **50**, 17953–17979.
 - 42 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B*, 1976, **13**, 5188–5192.
 - 43 A. J. Hensley, R. Zhang, Y. Wang and J. S. McEwen, Tailoring the adsorption of benzene on PdFe surfaces: a density functional theory study, *J. Phys. Chem. C*, 2013, **117**(46), 24317–24328.
 - 44 D. Tozini, M. Forti, P. Gargano, P. R. Alonso and G. H. Rubiolo, Charge difference calculation in Fe/Fe₃O₄ interfaces from DFT results, *Procedia Mater. Sci.*, 2015, **9**, 612–618.
 - 45 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jonsson, Origin of the Overpotential for Oxygen Reduction at a Fuel-Cell Cathode, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.
 - 46 O. My Na, N. T. Xuan Huynh, P. T. Thi, V. Chihaia and D. N. Son, Mechanism and activity of the oxygen reduction reaction on WTe₂ transition metal dichalcogenide with Te vacancy, *RSC Adv.*, 2020, **10**, 8460–8469.
 - 47 H. T. Thanh, O. K. Le, V. Chihaia and D. N. Son, Carbon dioxide conversion to methanol on a PdCo bimetallic catalyst, *Phys. Chem. Chem. Phys.*, 2024, **26**, 3963–3973.
 - 48 D. L. Quan, V. Chihaia and D. N. Son, Mechanisms and selectivity of methanol oxidation on PtRuM₃/C-MWCNT (M = Fe and Co) electrocatalysts, *RSC Adv.*, 2024, **14**, 25918–25931.
 - 49 J. Greeley and J. K. Nørskov, Electrochemical dissolution of surface alloys in acids: Thermodynamic trends from first-principles calculations, *Electrochim. Acta*, 2007, **52**(19), 5829–5836.
 - 50 W. M. Haynes, *CRC Handbook of Chemistry and Physics*, CRC Press, New York, 95th edn, 2014, pp. 5–80.
 - 51 W. Pepperhoff and M. Acet, The structure of iron, in *Constitution and Magnetism of Iron and its Alloys, Engineering Materials*, Springer, Berlin, Heidelberg, 2001, pp. 1–13.
 - 52 C. S. Yoo, P. Söderlind and H. Cynn, The phase diagram of cobalt at high pressure and temperature: the stability of cobalt and new-cobalt, *J. Phys.: Condens. Matter*, 1988, **10**(20), L311–L318.
 - 53 N. N. Lathiotakis, A. N. Andriotis, M. Menon and J. Connolly, Tight binding molecular dynamics study of Ni clusters, *J. Chem. Phys.*, 1996, **104**, 992–1003.
 - 54 G. Di Liberto, L. Giordano and G. Pacchioni, Predicting the Stability of Single-Atom Catalysts in Electrochemical Reactions, *ACS Catal.*, 2024, **14**(1), 45–55.

