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Potassium iodide-doped cesium tin chloride/copper oxide perovskite for light photodetection and energy conversion

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Perovskite materials have emerged as up-and-coming candidates for optoelectronic devices. This study developed a photodetector by depositing a halide double perovskite material, KI-doped Cs₂SnCl₆ (KI:CTC), onto a CuO substrate. Extensive analyses examined the photodetector's optical, structural, and morphological properties. The bilayer structure of KI:CTC exhibited a well-defined cubic crystal structure with a crystal size of 37.73 nm. The photodetector's band gap (E_g) was accurately measured to be 1.55 eV. The performance parameters of the KI:CTC/CuO photodetector were rigorously evaluated. At an incident photon wavelength of 390 nm, a remarkable photoresponsivity (PR) of 143.0 mA W⁻¹, an impressive specific detectivity (D*) of 7.95 × 10⁹ Jones, and a high external quantum efficiency (EQE) of 61.0% were achieved. Additionally, the stability and underlying mechanism of the photodetector were thoroughly discussed. The results suggest that the KI:CTC/CuO photodetector exhibits effective light detection, positioning it as a promising candidate for future optoelectronic applications.

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1. Introduction

Photodetectors (PDs) play a crucial role in converting light into electrical signals. They are used in optical communications, video imaging, security, biomedical imaging, motion detection, and gas sensing. Metal halide double perovskites are highly valued in optoelectronic applications due to their exceptional properties.^{1–4} Solar energy conversion through light absorption in semiconductor structures with photo-generated hot carriers is highly important for addressing the challenges of energy depletion.^{5,6} Among them, Cs₂SnCl₆ double halide perovskite has emerged as a favorable material for optoelectronics and solar cell technology.⁷ It exhibits outstanding optical properties, including a high dielectric constant, excellent optical absorption, a high static refractive index, good conductivity, and low reflectivity.^{7,8} The band gap of Cs₂SnCl₆ perovskites corresponds to the first empty conduction band, involving a transition from Cl-p to Sn-p orbitals. In that approach, the density of states predicted the Cs₂SnCl₆ behavior as p-type semiconductor.⁸ Additionally, Cs₂SnCl₆ demonstrates remarkable thermal, thermodynamic, and dynamic stability. It enables efficient

charge carrier transport and captures a wide range of light. The non-toxic nature and high Seebeck coefficient of Cs₂SnCl₆ further enhance its attractiveness for applications in thermoelectric devices and spintronic technology.

On the other hand, CuO possesses several notable characteristics, including a high melting temperature, electron correlation effect, spin dynamics, and conductivity.⁹ It functions as an n-type semiconductor material with a narrow bandgap. CuO-based photodetectors have gained significant attention because their ease of fabrication, affordability, and mechanical stability.^{10,11}

Desirable performance characteristics for photodetectors encompass high sensitivity, efficiency, and safety. While previous research has explored various aspects of photodetector development, significant limitations remain. Wang *et al.* fabricated CuO nanowires that produced a modest photocurrent of 20 μA under a high applied bias of +5 V, restricting their practical application.¹² Yinghua *et al.* integrated CuO with ZnO, achieving only slight improvements, with a photocurrent density (J_{ph}) of 107 μA at +1 V.¹³ Similarly, Qingshui *et al.* observed a low photocurrent of 12 μA in CuO/Si-based photodetectors, further limiting their viability.¹⁴ Additionally, Saidaminov *et al.* employed toxic lead-based materials in Au/CsPbBr₃/Pt photodetectors, raising significant health and environmental concerns.¹⁵ Zhuowei *et al.* reported a photoresponsivity of 48 mA W⁻¹ for Cs₂AgBiBr₆, while Ziqing *et al.* found Cs₃Bi₂Br₆/Cs₃BiBr₆ to exhibit a notably low photoresponsivity of 5.6 mA W⁻¹, limiting sensitivity and effectiveness.^{16,17} These studies reveal a gap between current photodetector performance and

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the desired standards, emphasizing the need for further advancements to achieve more optimal outcomes.

The integration of KI:CTC perovskite with CuO enables the development of a high-performance photodetector. This improvement arises from the synergistic interaction between the KI:CTC and CuO layers, as well as the formation of a KI:CTC/CuO pn junction, which enhances charge separation and transport.^{8,18} In this study, a novel KI:CTC/CuO pn junction was introduced as an efficient photodetector. The materials were characterized using various techniques, including X-ray diffraction, scanning electron microscopy, and UV-visible spectroscopy. The device's performance was evaluated under different wavelengths and light intensities, with key photodetector parameters assessed. The synergistic effects between KI:CTC and CuO significantly improved the photodetector's performance, leading to a photodetector with a broad spectral response, outstanding detectivity, high responsivity, fast response time, and excellent photo-stability. The exceptional optical properties of KI:CTC, combined with the robust stability of CuO, make this photodetector particularly well-suited for a wide range of optoelectronic applications, offering a promising alternative to conventional photodetectors.

2. Experimental

2.1. Preparation of CuO layer

A high-purity commercial Cu foil (El-Nasr Co. in Cairo, Egypt) undergoes a cleaning process. The process initiates with the polishing of the Cu foil using concentrated H₂SO₄ for 20.0 minutes. Subsequently, it is cleaned with acetone in an ultrasonic bath, followed by rinsing with distilled H₂O for 30.0 minutes. The thoroughly cleaned Cu foil is then immersed in a solution containing 0.2 M Fe(NO₃)₂ for 50 minutes at 70.0 °C. Lastly, the Cu sheet is annealed in atmospheric air for 15.0 minutes at a temperature of 500.0 °C, forming a CuO layer.

2.2. Synthesis of KI:CTC/CuO bilayer

To prepare the KI:CTC (potassium iodide-doped cesium tin chloride) solution, an optimized concentration of 0.01 M : 0.01

M : 0.05 M of tin chloride (SnCl₂), cesium chloride (CsCl), and potassium iodide (KI), respectively, is dissolved in 4 ml of dimethylformamide (C₃H₇NO). This ratio was determined after many tries to achieve high-quality, uniform films with excellent adhesion to the substrate. In this reaction, tin is oxidized from the +2 oxidation state in SnCl₂ to the +4 oxidation state in Cs₂SnCl₆. This oxidation process is facilitated by the presence of iodide ions (I⁻) from KI, which initially reduce Sn⁴⁺ to Sn²⁺. Subsequently, chlorine ions (Cl⁻) re-oxidize tin back to the Sn⁴⁺ state.^{19,20}

All the chemicals were obtained from Sigma Aldrich (USA) and have a purity level of 98.0%. The resulting KICsSnCl solution is then stirred magnetically at 100 rpm and a temperature of 50 °C for two hours. Approximately 140 μl of the KICsSnCl solution is placed onto the CuO surface at a temperature of 70.0 °C for 20.0 minutes to fabricate KI:CTC/CuO bilayers. Subsequently, the KI:CTC/CuO bilayers are dried at 130.0 °C for 15.0 minutes to complete the preparation process.

2.3. Characterization

The prepared layers were analyzed for their morphologies using a Zeiss Auriga 60 scanning electron microscope (SEM) working at an accelerating voltage of 5 keV. To identify the crystal phases present in the layers, X-ray diffraction (XRD) analysis was made by using a D5000 Siemens/Bruker X-ray diffractometer, with CuKα radiation. The optical properties were examined in the range of 200–2000 nm using an Elmer Perkin Lambda 950 spectrophotometer from the USA.

2.4. Photodetector process

The performance of the KI:CTC/CuO photodetector was evaluated using a workstation (CHI660E Instruments, China). The measurement setup is illustrated in Fig. 1. A xenon arc lamp (Newport Company, UK) served as the light source. The photodetector was fabricated by placing two electrodes on the surface of the KI:CTC/CuO using silver paste, with an active device area of 1.0 × 1.0 cm². The two Ag adhesive electrodes serve solely to collect free carriers generated in the KI:CTC/CuO bilayer and

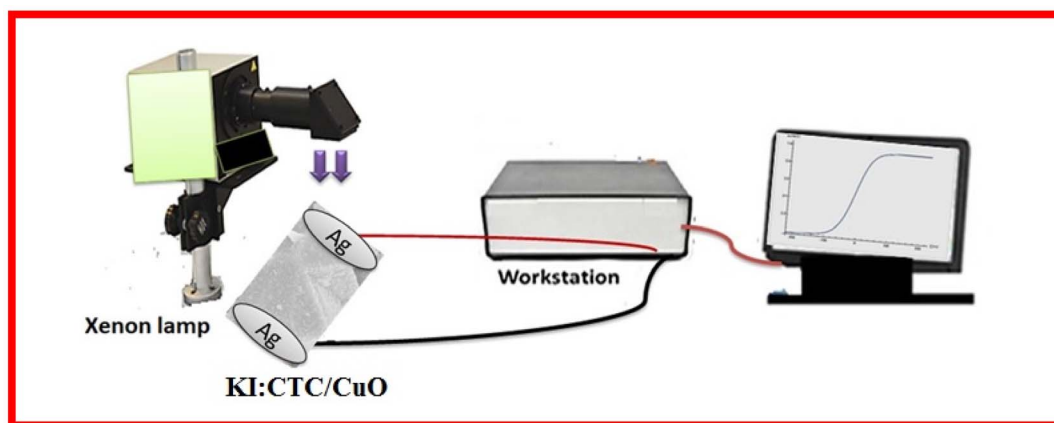


Fig. 1 Schematic diagram of the testing setup for the prepared KI:CTC/CuO photodetector.



facilitate their flow into the external circuit. These Ag electrodes are designed to be non-intrusive; they do not influence the photoresponse mechanisms of charge generation, transfer, or recombination within the photodetector material. The photodetector's performance was analyzed under varying white light power intensities ($25.0\text{--}100.0\text{ mW cm}^{-2}$) and incident light wavelengths ($390.0\text{--}636.0\text{ nm}$) at a fixed intensity of 75 mW cm^{-2} . Measurements were conducted in ambient air at room temperature, with an applied voltage range of -1.0 to $+1.0\text{ V}$. Additionally, the stability of the device was tested under continuous light exposure.

3. Results and discussion

3.1. Surface morphology

The surface morphology of the photoelectrode plays a crucial role in its functionality and effectiveness. Fig. 2(a) presents a SEM image of the CuO layer grown on the Cu foil. The image shows the formation of micro-stone structures composed of aggregated particles with varying shapes. The size of the CuO particles ranges from 40 to 1100 nm, indicating dimensional variations within the structure. To further investigate the surface features, a high-magnification SEM image is provided in Fig. 2(b), enabling a detailed examination. Surface roughness analysis was performed using ImageJ software, as depicted in Fig. 2(c). The 3D image of the roughness highlights the presence of valleys and peaks in the CuO surface topography, contributing to a rough surface texture. The roughness parameters quantify surface texture irregularities, providing valuable insights into material characteristics. Various metrics are used for this assessment. The arithmetic average roughness (R_a) represents the mean surface roughness over the entire measurement area, while the root mean square roughness (R_q) indicates the height variation across the surface.²¹ CuO exhibits R_a of 0.376 and R_q of 0.153.

Fig. 3 presents the SEM image of the KI:CTC layer deposited on CuO. The image reveals a compact surface morphology of the

KI:CTC layer, effectively covering the CuO layer. Additionally, irregular long structures in random directions, needle-like crystals, and small particles are observed, indicating the presence of different particle types within the material. At higher magnifications in Fig. 3(b), these features become more pronounced.

The surface topography depicted in Fig. 3(c) exhibits a rough and uneven texture, indicative of a heterogeneous microstructure. The KI:CTC/CuO sample displays R_a of 0.388 and a R_q of 0.173. These higher roughness parameters for KI:CTC/CuO suggest an enhanced surface texture when compared to CuO. The rough surface increases the surface area available for light absorption, enhancing the interaction between incident photons and the active material. It acts as a light-trapping structure, increasing the path length of photons and promoting absorption. Additionally, the rough surface reduces light reflection and increases light scattering, ensuring more efficient utilization of incoming photons. The surface irregularities also facilitate effective charge separation and collection, resulting in improved sensitivity and overall performance of the photodetector.

3.2. X-ray diffraction (XRD) measurements

The XRD analysis gives valuable information on the structural characteristics of the fabricated layers, contributing to the understanding of photodetector performance. Fig. 4 (black color) illustrates the XRD analysis results for the CuO layer. According to JCPDS cards #96-900-7498 and #96-101-1195, the analysis reveals the presence of two mixed phases: CuO tenorite (monoclinic) and Cu_2O cuprite (cubic).^{22–25} The dominant phase peaks belong to CuO. The peaks of CuO are observed at 2θ angles of 61.652° , 38.75° , and 35.56° corresponding to the Miller crystallographic planes $[-113]$, $[111]$, and $[-111]$, respectively. Likewise, the Cu_2O peaks are observed at 2θ angles of 42.35° and 65.92° , corresponding to the $[020]$ and $[112]$ planes, respectively. Using the Debye–Scherrer equation, the

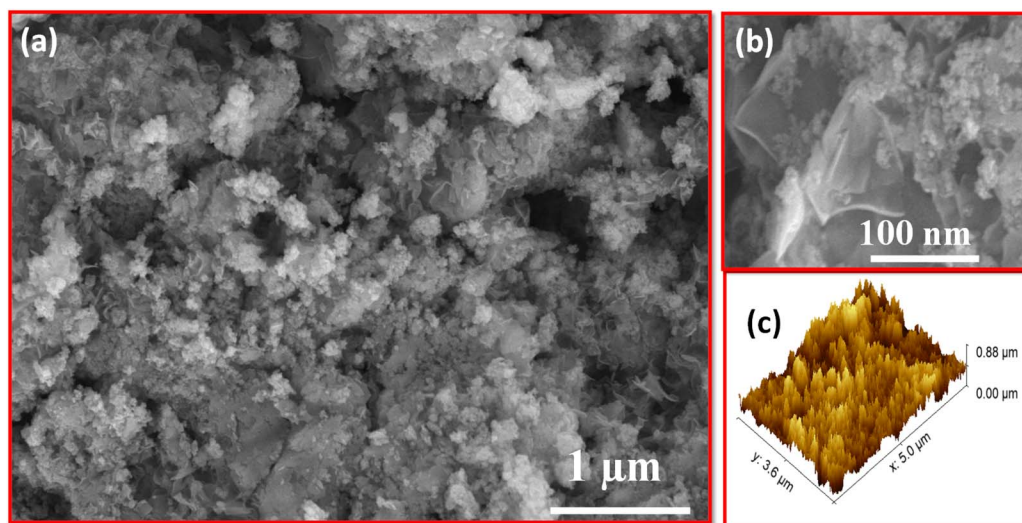


Fig. 2 (a and b) SEM images with two magnifications and (c) is the roughness 3D image for CuO layer.



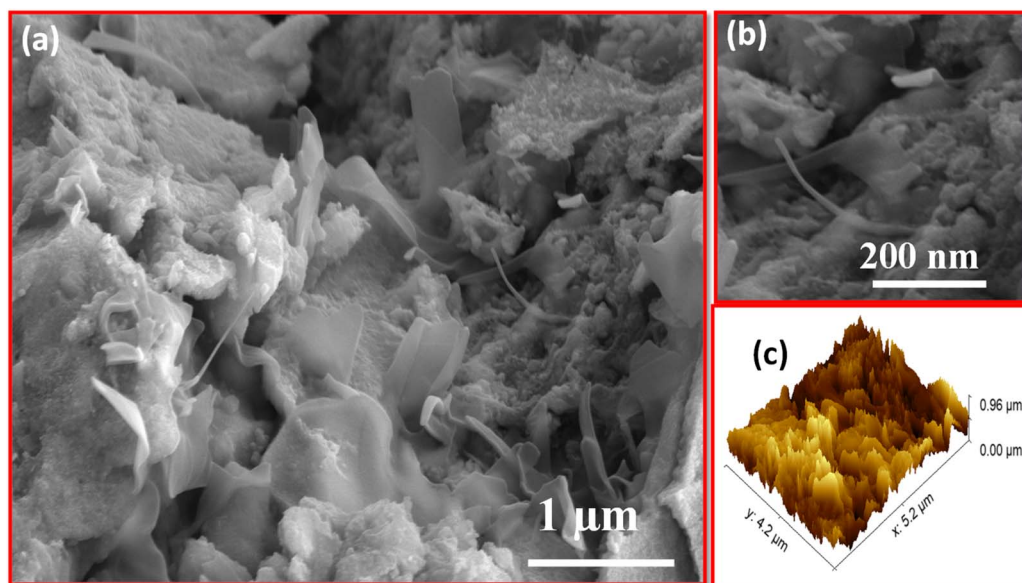


Fig. 3 (a and b) Different SEM magnifications and (c) is the roughness 3D image of KI:CTC/CuO bilayers.

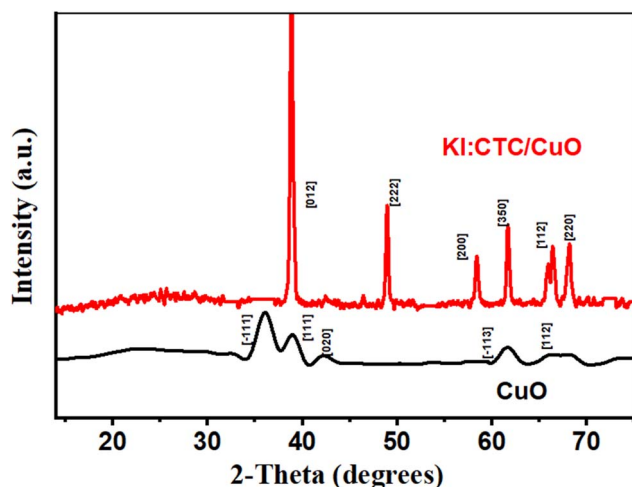


Fig. 4 XRD of CuO and KI:CTC/CuO films.

crystal sizes are estimated to be approximately 34.75 nm for CuO and 28.31 nm for Cu₂O.

The XRD spectrum of the KI:CTC/CuO bilayers is depicted in Fig. 4 (red curve). The presence of narrow, highly intense peaks in the KI:CTC/CuO spectrum indicates enhanced crystallinity of the bilayer structure. According to JCPDS cards #07-0197 and #75-0376, Cs₂SnCl₆ crystallizes in a cubic structure with the *Fm*3*m* space group, where Sn adopts the +4 oxidation state.^{26–29} Distinct XRD peaks corresponding to Cs₂SnCl₆ are observed at 2θ angles of 58.31°, 39.07°, 68.01°, and 66.28°, indexed to the [200], [012], [220], and [112] crystallographic planes, respectively.^{26–29}

Furthermore, Fig. 4(b) presents the XRD data, which reveal the orthorhombic structure of KI, with peaks observed at 2θ angles of 61.45° and 48.83°, corresponding to the [350] and [222] planes, respectively. This agrees with card #01-0796,

confirming the successful doping of KI into CTC.^{30,31} The crystal size of KI:CTC/CuO is estimated to be approximately 37.73 nm. The high-quality crystal structure of the bilayers in the photo-detector plays a significant role in enhancing its stability and durability.

3.3. Optical analysis

Fig. 5 (black color) presents an analysis of the optical properties of the CuO layer. The figure displays several optical parameters, including absorbance (*A*), transmittance (*T*), reflectance (*R*), and band gap (*E_g*). In Fig. 5(a), the absorbance curve of the CuO layer shows a prominent peak. This peak is located in the UV-visible range. It arises from electronic photoexcitation from the valence band (VB) to the conduction band (CB). Notably, the CuO absorption rapidly between 400 and 520 nm and reaches a steady state. Additionally, the absorption peak at 840 nm can be attributed to the increased thickness of the CuO layer, resulting in extended absorption into the near-infrared region.^{32,33}

Fig. 5(b) demonstrates that the transmittance of CuO is initially close to zero for wavelengths exceeding 430 nm. Subsequently, there is an exponential increase in transmittance with wavelength until it reaches a plateau state at $\lambda > 600$ nm. The reflectance spectrum in Fig. 5(c) shows low values, indicating significant absorption of the CuO layer in the UV-visible regions. Additionally, the reflectance spectrum displays oscillating interference fringes associated with the CuO layer.³⁴

The absorption coefficient (α) can be determined using the film thickness (*d*) and absorbance (*A*), as expressed in the next equation³⁵

$$\alpha = 2.3(A/d) \quad (1)$$

The optical band gap (*E_g*) can be calculated by using Tauc model



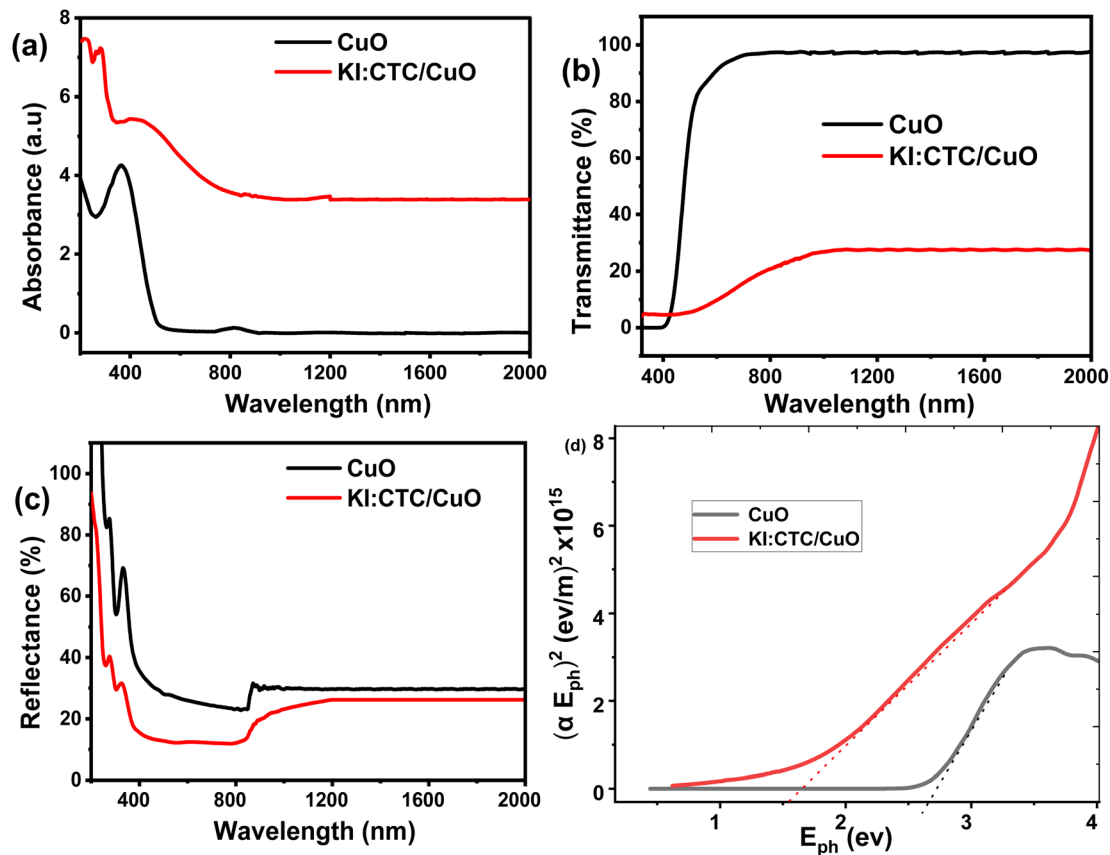


Fig. 5 Optical properties of CuO and KI:CTC/CuO: (a) absorbance, (b) transmittance, (c) reflectance, and (d) band gap.

$$E_{\text{ph}}\alpha = M\sqrt{(E_{\text{ph}} - E_{\text{g}})} \quad (2)$$

In eqn (2), M represents a constant, and E_{ph} denotes the energy of the incident light. Fig. 5(d) shows that the band gap of the CuO layer is measured to be 2.67 eV. The relatively high value of the CuO band gap compared to those reported in other works may be attributed to the existence of the Cu_2O phase, as observed in the XRD spectrum.

The optical properties of a bilayer composed of KI:CTC/CuO are discussed in Fig. 5 (red color). In Fig. 5(a), the absorbance measurements demonstrate that the KI:CTC/CuO bilayer efficiently absorbs light in the 400 to 800 nm wavelength range. Consequently, this bilayer exhibits a redshift towards the visible light spectrum, showing higher absorbance compared to the CuO film. This indicates an increase in photon absorption, leading to enhanced production of charge carriers.³⁶ Fig. 5(b) demonstrates the transmittance characteristics of the bilayer, showing close to zero transmittance for wavelengths shorter than $\lambda > 400$ nm and exponential growth as wavelength increases. Beyond $\lambda > 600$ nm, the transmittance reaches a plateau. This indicates high opacity to UV and visible light and increasing transparency at longer wavelengths. Fig. 5(c) shows that the KI:CTC/CuO bilayer has lower reflectance than the CuO layer, particularly in the UV-visible regions. The reduced reflectance is attributed to effective light modulation at the interfaces of the KI:CTC layer.³⁷

Fig. 4(d) provides information about the band gap of the KI:CTC/CuO, which is measured at 1.55 eV, indicating its suitability for various optoelectronic applications due to this relatively low band gap. While materials with larger band gaps primarily absorb light in the UV spectrum, which can limit their effectiveness in detecting visible or infrared light. The high-band-gap materials have significant applications in UV photo-detection.³⁸ These include environmental monitoring, UV sterilization, and detecting emissions from UV sources like LEDs and lamps.³⁹ Additionally, they are crucial in health and safety applications, such as monitoring UV exposure to prevent skin damage.⁴⁰ Furthermore, these materials are vital for space-based applications, contributing to studies of solar activity and its effects on Earth's atmosphere.⁴¹ The development of sensors utilizing these materials can enhance photonic devices for UV imaging and spectrometry.⁴²

The optical bandgap decreases when CuO is combined with KI:CTC, as seen in Fig. 5(d). This can be attributed to band structure modification and interfacial interactions between the two materials. The formation of a heterojunction leads to energy level alignment, which reduces the overall optical bandgap. Additionally, charge transfer effects and localized states at the interface may introduce intermediate energy states within the band structure, further lowering the effective bandgap. Furthermore, the increased surface roughness enhances light interaction with the KI:CTC/CuO structure by



expanding the surface area, resulting in greater light scattering and trapping, ultimately improving the material's optical properties. As a result, the efficient light absorption and relatively low bandgap value of KI:CTC/CuO indicate the potential of the KI:CTC/CuO bilayer in photodetector applications.

3.4. KI:CTC/CuO photodetector

3.4.1. Influence of light power intensity. Fig. 6 illustrates the photocurrent (J_{ph}) of KI:CTC/CuO bilayers as a function of applied voltages under various white light power intensities. The voltage ranges from -1.0 to $+1.0$ V, and the light power intensities range from 25.0 to 100.0 mW cm^{-2} . In the absence of light (dark condition), Fig. 6(a) shows that the dark current (J_d) changes from -0.23 to 0.20 mA cm^{-2} as the voltage varies from -1 to $+1$ V. This observation indicates that in the dark condition, only a small number of free electrons in the KI:CTC/CuO bilayers exhibit a response to the applied potential. The dark current is primarily generated by thermally induced carriers and electrical leakage. In the dark conditions, O_2 molecules from the air adsorb onto the surface of KI:CTC/CuO bilayers. The adsorption process takes place as a result of the presence of active sites or chemical interactions on the surface of the bilayer. The adsorbed O_2 molecules can capture the free electrons present on the surface of the KI:CTC/CuO perovskite bilayer. As a result, the resistivity of the bilayer increases, leading to a low dark current density.^{43,44}

With an increase in power intensity of incident white light (from 25.0 to 100.0 mW cm^{-2}), the photocurrent (J_{ph}) of the KI:CTC/CuO bilayers also increases. The photo-excited current is 67.5 times higher than the dark current at a fixed bias of 1 V. The values of photocurrent are high enough for direct measurement without any amplifier. When the bilayers are illuminated, the electrons absorb the energy from the incident light and transition from VB to CB. This process creates electron-hole pairs (e-h). The photo-generated carriers are efficiently separated due to the presence of an applied bias voltage. The holes tend to migrate towards the surface, where they interact with the adsorbed oxygen ions. Hence, holes remove

negative charges and discharge the oxygen ions. So, the desorption of O_2 ions is facilitated by the presence of holes. Consequently, the conductivity of the KI:CTC/CuO bilayers improves. This leads to a reduction in resistance and an increase in photocurrent under light illumination. The photocurrent rises as the applied voltage increases due to the faster drift velocity and increase in photogenerated electron-hole pair separation.

Fig. 6(b) presents the relationship between photocurrent density (J_{ph}) and light power intensity (P) at $+1.0$ V for the KI:CTC/CuO bilayers. The photocurrent density increases from 8.66 to 13.56 mA cm^{-2} as the power intensity rises from 25.0 to 100.0 mW cm^{-2} . This demonstrates the strong dependence of the generated photocurrent on the light power intensity. With an increase in light intensity, more electrons within the KI:CTC/CuO bilayers are excited from VB to CB. Consequently, additional charge carriers (electron-hole pairs) are generated on the surface of the material. This increased availability of carriers contributes to the observed higher photocurrent.

It is important to note that the relationship between incident light intensity and the resulting photocurrent density is nonlinear. This nonlinearity indicates the complexity of the electron-hole transportation reaction occurring in the KI:CTC/CuO bilayers. Several factors may influence the overall photocurrent response, such as carrier recombination, trapping, or other transport mechanisms. These factors can affect the efficiency of charge carrier generation, transport, and collection, leading to the photocurrent density does not increase proportionally with an increase in incident light intensity.⁴⁵

The relationship between J_{ph} and P for the KI:CTC/CuO bilayers can be described using the power-law equation

$$J_{ph} = CP^k \quad (3)$$

Here, C represents constant and P represents the incident light power. In the context of photodetectors, the exponent parameter (k) reflects processes such as e/h generation, recombination, and carrier trapping.⁴⁶ When fitting the experimental results in Fig. 6(b) with eqn (3), an approximate exponent

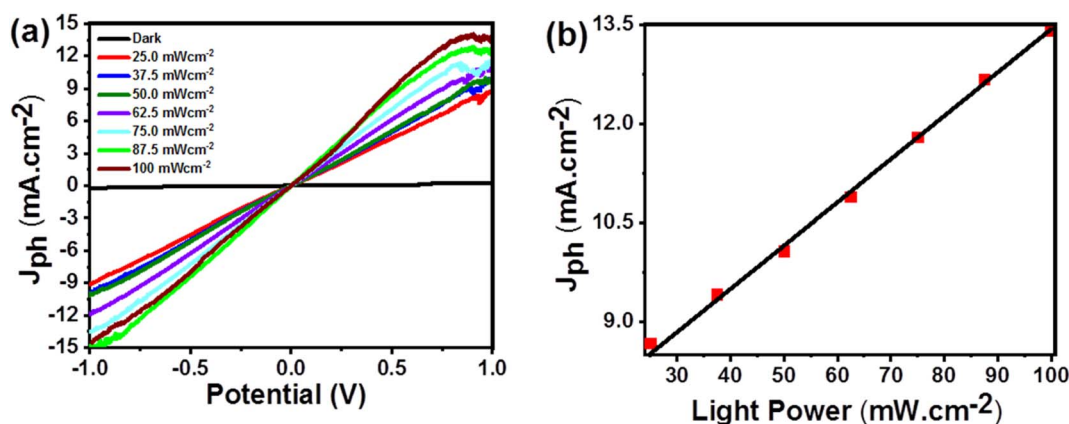


Fig. 6 (a) Effect of the light power density on the photocurrent values (b) the relationship between the photocurrent values and the light power at $+1$ V for the KI:CTC/CuO photodetector.



parameter (k) of 0.96 is observed. This indicates the ability of KI:CTC/CuO bilayers to generate and effectively separate photoinduced charge carriers in response to incident light. Consequently, the bilayer material can efficiently convert changes in light power into proportional changes in photocurrent density. As a result, the bilayer material exhibits excellent photosensing ability, rendering it suitable for use as a photodetector.

3.4.2. Light wavelength. The correlation between the wavelength of light and the corresponding photocurrent values holds significant importance. To investigate the spectral response of the KI:CTC/CuO bilayer, photocurrent–voltage ($J_{\text{ph}}-V$) measurements were performed in dark condition and across a range of monochromatic wavelengths spanning the UV-visible regions as display in Fig. 7(a). In the absence of light (dark conditions), Fig. 7(a) shows that the dark current density (J_{d}) varies from -0.15 to 0.10 mA cm^{-2} as the applied voltage sweeps from -1.0 to $+1.0$ V. This observation suggests that under dark conditions, only a limited number of free electrons in the KI:CTC/CuO bilayers respond to the applied voltage. The O_2 molecules from the atmosphere adsorb onto the surface of the KI:CTC/CuO bilayers in the absence of light. The adsorption occurs due to the presence of active sites or chemical interactions on the bilayer's surface. The adsorbed O_2 molecules capture free electrons located on the surface of the perovskite bilayer. Consequently, this leads to an increase in the resistivity of the bilayer, resulting in a low dark current density.^{43,44}

The photocurrent–voltage measurement across a range of monochromatic wavelengths is presented in Fig. 7(a). This analysis covers wavelengths from 390 to 636 nm, with the light intensity for all different wavelengths maintained at a constant 75.0 mW cm^{-2} . As the wavelengths increase, the J_{ph} values decrease. Fig. 7(b) further illustrates this correlation by plotting the photocurrent values at $+1$ V against different monochromatic light sources during illumination. According to Fig. 7(b), the photocurrent density decreases non-linearly as the incident light wavelength increases. Specifically, the photocurrent density at a wavelength of 390.0 nm is 12.8 mA cm^{-2} , while at 636.0 nm, it drops to 3.4 mA cm^{-2} . This decrease in

photocurrent is due to the longer wavelengths of light carry lower energy photons. These lower energy photons possess a reduced ability to excite electrons across the energy bandgap of the material. Consequently, the diminished excitation of electrons leads to a reduction in photocurrent. The high photocurrent observed under short-wavelength light irradiation in the UV region attributed to the increased generation of e/h pairs. This occurs when the photon energy exceeds the bandgap of the bilayer, leading to a greater number of photo-generated carriers. This indicates the photoresponse strongly depends on the photon wavelength. This proves the broadband detection capability from ultraviolet to the visible range.

3.4.3. Performance analysis. The performance of a photodetector is commonly assessed by considering several parameters, namely the photoresponsivity, specific detectivity, and external quantum efficiency. These parameters provide valuable insights into various aspects of the detector's performance.⁴⁷

The photoresponsivity (PR) is a crucial factor for evaluating the performance of a photodetector. PR denotes the ratio of the generated photocurrent to the incident light power on the effective area of the photodetector. The calculation is performed using the following equation.^{48,49}

$$\text{PR} = \frac{(J_{\text{ph}} - J_{\text{d}})}{\text{light power}} \quad (5)$$

The photoresponsivity is determined by analyzing the relationship between light intensity and photocurrent density at a specific applied voltage. Fig. 8(a) displays the PR values for different applied wavelengths at $+1$ V. Notably, the maximum PR of 143 mA W^{-1} is observed at 390 nm, which agrees with the J_{ph} results depicted in Fig. 7(b). The PR is lower in the visible region compared to the UV region due to factors such as reduced absorption and shorter recombination time in that range.

External quantum efficiency (EQE) is a key factor that measures the efficiency of a photodetector in converting incident light photon flux into photocurrent.^{50,51} The flux of photons is directly proportional to the intensity of the incident

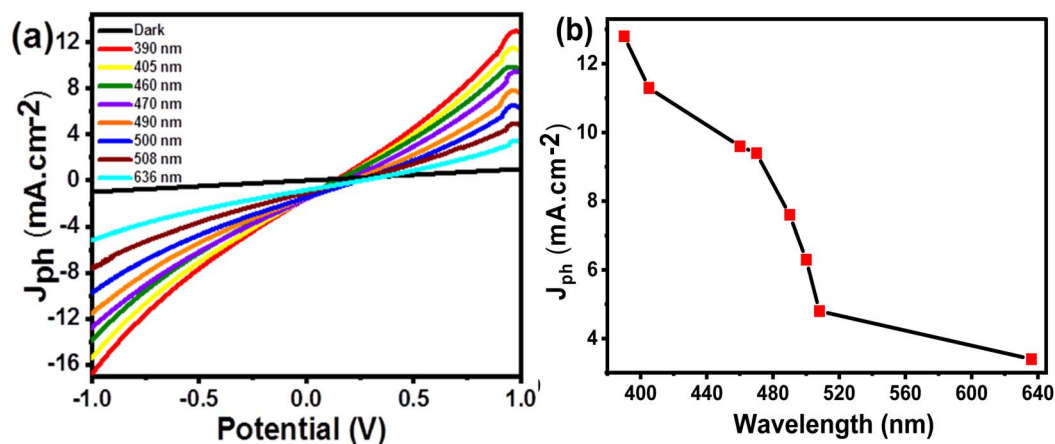


Fig. 7 (a) The relationship between the photocurrent values and the voltage in the presence of several wavelengths and (b) the correlation of $J_{\text{ph}} - \lambda$ at $+1$ V for KI:CTC/CuO photodetector.

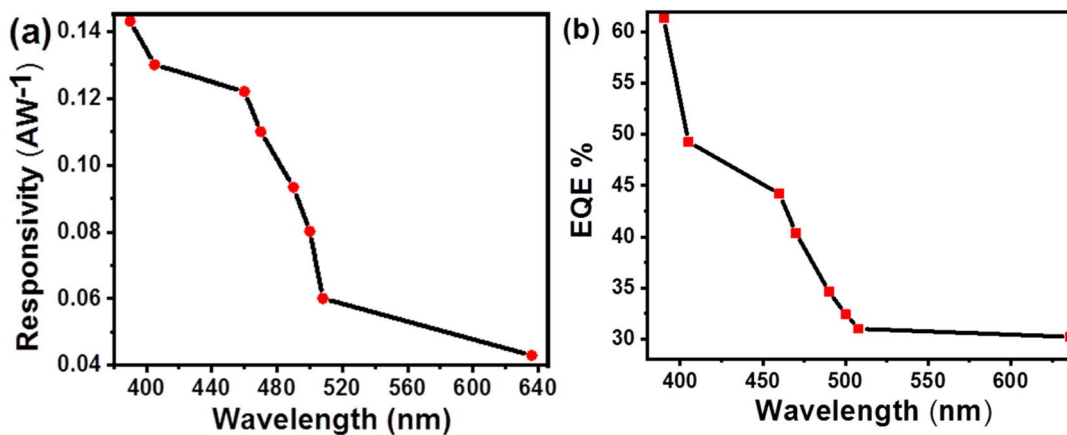


Fig. 8 (a) The relationship between photoresponsivity–wavelengths and (b) the EQE under different intensities of light illumination for the KI:CTC/CuO photodetector.

light. EQE is defined as the ratio of the number of generated charge carriers to the number of incident photons. The bandgap and absorption coefficient of the photodetector material influence its EQE. A higher EQE value corresponds to a more efficient photodetector that can convert a larger fraction of incident photons into detectable charge carriers. High EQE generally leads to higher D^* . The EQE can be determined using the next equation⁵²

$$\text{EQE} = \text{PR} \left(\frac{1240}{\lambda} \right) 100 \quad (7)$$

Based on Fig. 8(b), it can be observed that the external quantum efficiency (EQE) of the KI:CTC/CuO perovskite bilayer decreases from 61.0% to 30.0% as the wavelength increases from 390.0 nm to 636.0 nm. This trend indicates a reduction in the efficiency of converting incident photons into electric current as the wavelength of light increases, which agrees with the responsivity (PR) values in Fig. 8(a).

Detectivity (D^*) is a valuable parameter used for comparing the performance of different photodetectors under specific conditions, such as a particular wavelength and bandwidth. It quantifies the minimum detectable signal that the photodetector can reliably differentiate from the noise level. This characteristic is important for applications that require the detection of weak light signals in the presence of background noise, such as in astronomy, night vision systems, medical imaging, optical communication, and sensing. The detectivity can be calculated using eqn (6) (ref. 49)

$$D^* = \text{PR} \sqrt{\frac{A}{2eJ_d}} \quad (6)$$

Here, A represents the surface area of the photodetector, and e represents the electron charge. The specific detectivity reaches its maximum value of $D^* = 7.95 \times 10^9$ Jones at a wavelength of 390 nm. A higher detectivity value indicates a more sensitive photodetector that can detect weaker signals with greater accuracy.

3.4.4. Stability and reproducibility. Reproducibility is indeed a vital characteristic of photodetectors as it ensures

accurate and reliable data collection across repeated measurements. A highly reproducible photodetector maintains consistent readings without variations over multiple measurement cycles under identical conditions. This aspect is especially critical in industrial applications. To assess the reproducibility of the KI:CTC/CuO photodetector, J_{ph} -voltage measurements were performed multiple times at a light intensity of 100.0 mW cm^{-2} . Fig. 9(a) demonstrates that all runs have nearly identical values, indicating that the KI:CTC/CuO photodetector exhibits good reproducibility. This characteristic is important for the mass production and commercialization of photodetectors.

Stability refers to the photodetector's ability to maintain a consistent and reliable response over an extended duration, which is crucial for long-term performance and reliability. Factors such as aging of components, degradation of materials, operating conditions, and electronic drift can affect the stability. To assess the stability of the KI:CTC/CuO photodetector, chronoamperometry was performed. This technique involves applying a voltage of 1.0 V and measuring the photocurrent values under a light intensity of 100.0 mW cm^{-2} . By monitoring the photocurrent over an extended period, the stability of the photodetector can be evaluated. Analysis of Fig. 9(b) reveals that the photocurrent values initially decreased during the first 270 seconds, followed by a nearly constant level maintained until 500 seconds. These findings indicate an acceptable level of stability for the prepared KI:CTC/CuO photodetector. The presence of oxygen molecules adsorbed on the KI:CTC/CuO surface contributes to a relatively small change in the photocurrent values over time. The use of stable inorganic materials in the construction of the KI:CTC/CuO photodetector significantly contributes to its high stability.⁵³ The improved crystallinity and low defect density in the KI:CTC/CuO bilayers enhance the stability of photodetector applications' performance over an extended period. The high-quality crystal structure of the bilayers enhances charge transport properties, promotes efficient carrier generation, and reduces charge carrier recombination, resulting in stable photodetection capabilities. This also exhibits enhanced durability and decreases the chances of performance degradation or failure.



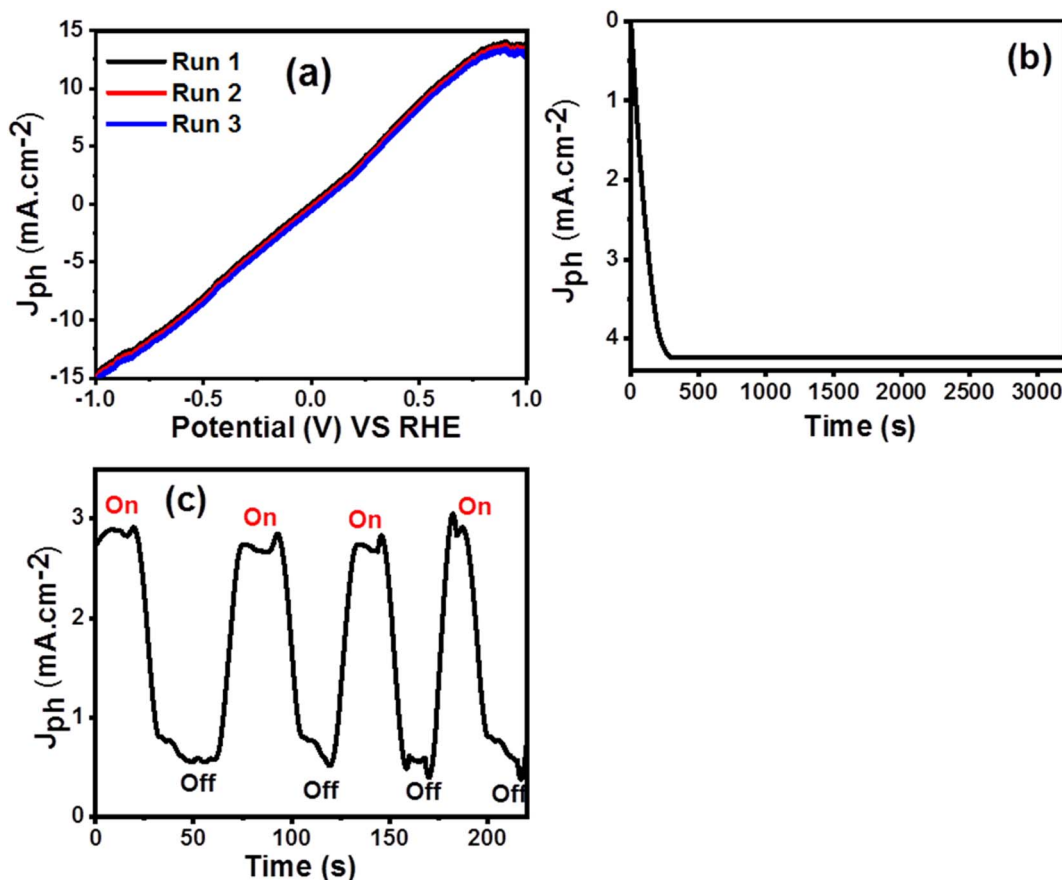


Fig. 9 (a) Reproducibility (b) stability, and (c) chopping current of the KI:CTC/CuO photodetector.

On/off light chopping is a technique used to achieve more precise measurement and analysis of the photodetector's performance by isolating the photocurrent generated by the incident light. This technique involves alternating between periods of white illumination provided by the xenon lamp (light-on at 100 mW cm⁻²) and darkness (light-off at 0 mW cm⁻²). By using this technique, the signal of interest (the photocurrent generated by the incident light) can be separated from other sources of current, such as dark current or background noise. The transient photocurrent response to on/off illumination is exhibited in Fig. 9(c), where three cycles were measured. Good repeatable photocurrent generation indicates that the photoresponse of the photodetector is reversible, and the photocurrent remains stable without degradation. When the light is turned on, the photocurrent rapidly increases to its maximum value and remains stable. Conversely, when the light is turned off, the photocurrent drastically drops back to its initial value.

The on/off ratio, defined by the photocurrent/dark current ($\text{on/off} = J_{ph}/J_{dark}$), is another factor parameter of the photodetector. A high on/off ratio is desirable for practical applications of KI:CTC/CuO as a high-quality photosensitive switch. At an applied voltage of +1.0 V, the photodetector exhibits an on/off (J_{photo}/J_{dark}) ratio of approximately 3.0. The rapid increase in current and the stable value achieved demonstrate fast charge transfer and reliable performance.

The rise and decay times are critical parameters for evaluating the response speed of the photodetector. During the illumination on/off cycle, the rise time and decay time are defined as the time required for the photocurrent to transition from 10% to 90% and 90% to 10%, respectively, of its stable value. The calculated rise time is 20.0 s, while the decay time is 34.0 s, enabling effective detection and analysis of light signals. The observed hysteresis and delayed response during the switching process result from the complex charge generation and transport dynamics during the on/off cycles, which do not follow a linear pathway. This behavior suggests energy retention within the material, potentially affecting measurement accuracy and precision.^{54,55}

As a result, the photodetector holds strong potential for practical applications involving the detection of many wavelengths and light power intensities. Its competitive characteristics make it a promising choice in the field. Lastly, Table 1 compares the performance of the presented KI:CTC/CuO photodetector with earlier research.

4. Mechanism

Fig. 10 illustrates the energy diagram of the KI:CTC/CuO photodetector system and the charge transfer mechanisms. When the semiconductor material (either KI:CTC or CuO) absorbs suitable photons, it excites electrons from the VB to the

Table 1 The performance of the fabricated photodetector KI:CTC/CuO with earlier articles

Structure	Wavelength (nm)	Bais (V)	R (A W ⁻¹)	D (Jone)
Cu ₂ O/ZnO ⁵⁶	596	0	0.288×10^{-3}	—
<i>n</i> -Cu ₂ O/ <i>p</i> -CuI ⁵⁷	465	0	0.25	—
TiO ₂ /CsSnI ₃ /P3HT ⁵⁸	350–1000	0	0.257	1.5×10^{10}
CsSnI ₃ (ref. 59)	940	0.1	0.054	3.85×10^5
GO/Cu ₂ O ⁶⁰	300	2	0.0005	1.0×10^6
Cs ₂ SnI ₆ (ref. 61)	—	3	—	—
CuO nanowires ¹²	390	5	—	—
CuO/Si Nanowire ¹⁴	405	0.2	3.8×10^{-3}	—
ZnO–CuO ⁶²	405	1	3×10^{-3}	—
CuFeO ₂ /CuO/Cu ⁶³	390	1	0.33	—
Cu/CuO NP ⁶⁴	390	1	6.2×10^{-3}	—
KI:CTC/CuO (this work)	400	1	0.143	7.95×10^9

CB, creating e/h pairs. The KI:CTC/CuO bilayer enhances light absorption and e/h pair generation due to its low band gap, rough surface texture, and large surface area. However, the recombination of e/h pairs can decrease carrier mobility and hinder their contribution to the photocurrent.

At the interface between the KI:CTC and CuO layers, charge transfer occurs. Carrier diffusion between CuO and KI:CTC continues until the Fermi energy levels (E_F) of both materials reach equilibrium. This results in band bending of the electronic structure and the formation of a depleted region at the KI:CTC/CuO junction. Consequently, a pn junction is formed between the *n*-type CuO and *p*-type Cs₂SnCl₆ (KI:CTC), creating a built-in electric field. Under the influence of this internal electric field, electrons from CuO migrate to the CB of KI:CTC, while holes generated in KI:CTC move to the VB of CuO. The

built-in electric field thus acts as a potential barrier that separates e/h pairs more quickly and effectively.⁶⁵ In addition, the interface between the Ag electrodes and the KI:CTC/CuO layer can be crucial for optimizing charge transfer. The work function of silver can significantly influence the energy barrier for carrier injection, helping to create a built-in electric field that efficiently separates and collects photogenerated charges.⁶⁵ The junction's ability to improve light absorption leads to the creation of additional free carriers.⁶⁶ Moreover, the internal electric field facilitates carrier transfer and electron accumulation on the surface of KI:CTC. Consequently, the KI:CTC/CuO structure demonstrates an improved photoresponse, resulting in an increased photocurrent value.

The efficiency of the KI:CTC/CuO photodetector can be further enhanced by incorporating potassium iodide (KI) into

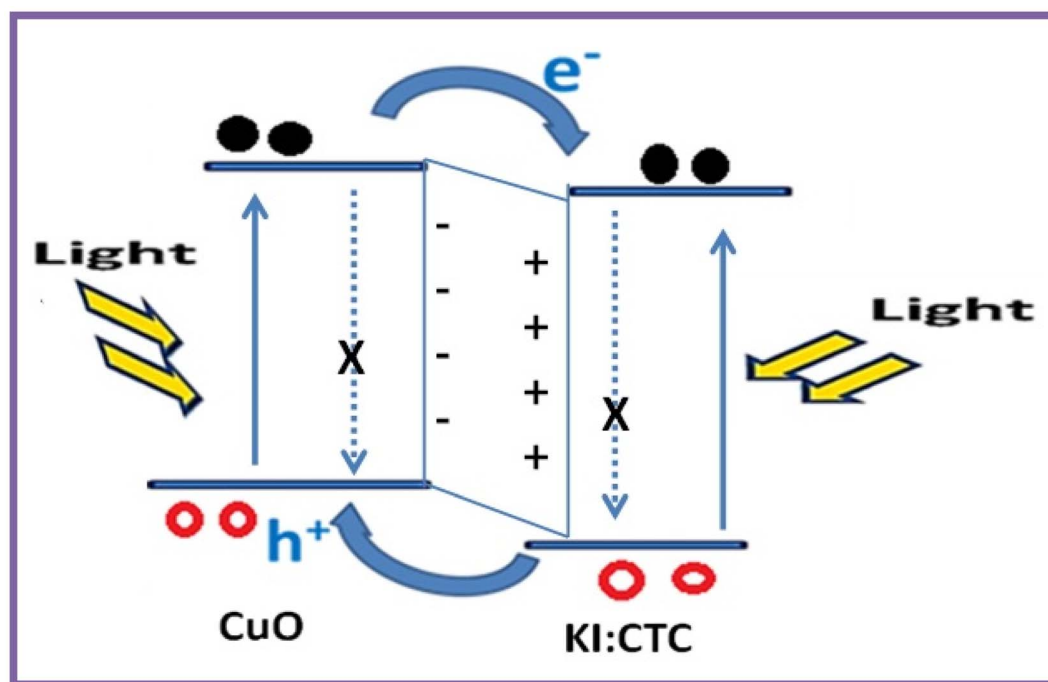


Fig. 10 Energy diagram of the KI:CTC/CuO photodetector system.



the system. While pure Cs_2SnCl_6 has its merits, it lacks the enhanced properties that KI brings to the table, such as improved absorption characteristics, better carrier mobility, and overall stability. KI modifies the band alignment of the perovskite material, reducing the charge-transfer barrier and improving conductivity.^{67,68} This enhancement facilitates more efficient charge transfer and higher photocurrent generation. Consequently, the photodetector exhibits high performance, making the KI:CTC/CuO photodetector well-suited for a variety of industrial applications.

5. Conclusion

In this study, a high-efficiency KI-doped Cs_2SnCl_6 (KI:CTC)/CuO bilayer was synthesized for photodetector applications. The KI:CTC/CuO bilayers were tested over a range of light intensities from 25.0 to 100.0 mW cm^{-2} and wavelengths from 390.0 to 636.0 nm. Various techniques were used to characterize the optical, chemical, and morphological properties of the bilayers. The KI:CTC/CuO bilayers exhibited good crystallinity and a low bandgap (E_g). Notably, the photoresponsivity of the bilayers was measured to be 143 mA W^{-1} , with an external quantum efficiency of 61.0% and a high detectivity of 7.95×10^9 Jones. The impressive performance demonstrated by the synthesized KI:CTC/CuO bilayer opens up exciting opportunities for its utilization across numerous fields and a wide range of technological applications.

Data availability

Data will be made available on request.

Author contributions

A. H. Ali: writing – original draft, methodology, formal analysis. A. M. Ahmed: formal analysis, data curation, software. S. M. Abass: investigation, visualization. A. A. Abdel-Khaliek: investigation, supervision. M. Shaban: resources, investigation. M. Al-Hmoud: data curation, visualization. S. Abd El Khalik: resources, visualization. M. Rabia: conceptualization, formal analysis, software.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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References

- 1 J. Park and K. Hong, Dual-site compositional engineering of bismuth-based halide perovskites for stable and efficient lead-free solar cells, *J. Phys. Chem. C*, 2021, **125**(24), 13138–13145.
- 2 F. Arjmand, Z. Golshani, S. Fatemi, S. Maghsoudi, A. Naeimi and S. Hosseini, The lead-free perovskite solar cells with the green synthesized BiI_3 and AgI nanoparticles using Vitex agnus-castus plant extract for HTM-free and carbon-based solar cells, *J. Mater. Res. Technol.*, 2022, **18**, 1922–1933.
- 3 S. Ghosh, H. Shankar and P. Kar, Recent developments of lead-free halide double perovskites: a new superstar in the optoelectronic field, *Mater. Adv.*, 2022, **3**(9), 3742–3765.
- 4 A. Rahman, et al., Enhancing the photo-response of CuO thin film based infrared light photodetector *via* fabricating ZnO-capped CuO device, *Phys. B*, 2025, **697**, 416713.
- 5 S. Boriskina, J. Zhou, W. Hsu, B. Liao and G. Chen, Limiting efficiencies of solar energy conversion and photo-detection *via* internal emission of hot electrons and hot holes in gold, *SPIE*, 2015, **9608**, 299–305.
- 6 D. Ninh, et al., Highly efficient photon energy conversion and ultrasensitive self-powered photodetection *via* a monolithic p-3C-SiC nanothin film on p-Si/n-Si double junction, *ACS Appl. Mater. Interfaces*, 2024, **16**(29), 38658–38668.
- 7 M. Hayat and R. Khalil, A DFT engineering of double halide type perovskites Cs_2SiCl_6 , Cs_2GeCl_6 , Cs_2SnCl_6 for optoelectronic applications, *Solid State Commun.*, 2023, **361**, 115064.
- 8 K. Bouferrache, et al., Crystal structure, mechanical, electronic, optical and thermoelectric characteristics of Cs_2MCl_6 (M= Se, Sn, Te and Ti) cubic double perovskites, *Results Phys.*, 2024, **56**, 107138.
- 9 I. Erdogan and O. Gullu, Optical and structural properties of CuO nanofilm: its diode application, *J. Alloys Compd.*, 2010, **492**(1–2), 378–383.
- 10 Q. Zhang, et al., CuO nanostructures: synthesis, characterization, growth mechanisms, fundamental properties, and applications, *Prog. Mater. Sci.*, 2014, **60**, 208–337.
- 11 K. Nomura, H. Ohta, A. Takagi, T. Kamiya, M. Hirano and H. Hosono, Room-temperature fabrication of transparent flexible thin-film transistors using amorphous oxide semiconductors, *Nature*, 2004, **432**(7016), 488–492.
- 12 S. Wang, et al., A CuO nanowire infrared photodetector, *Sens. Actuators, A*, 2011, **171**(2), 207–211.
- 13 Z. Bai and Y. Zhang, Self-powered UV–visible photodetectors based on ZnO/Cu₂O nanowire/electrolyte heterojunctions, *J. Alloys Compd.*, 2016, **675**, 325–330.
- 14 Q. Hong, Y. Cao, J. Xu, H. Lu, J. He and J. Sun, Self-powered ultrafast broadband photodetector based on p–n heterojunctions of CuO/Si nanowire array, *ACS Appl. Mater. Interfaces*, 2014, **6**(23), 20887–20894.
- 15 M. Saidaminov, et al., Inorganic lead halide perovskite single crystals: phase-selective low-temperature growth, carrier



- transport properties, and self-powered photodetection, *Adv. Opt. Mater.*, 2017, 5(2), 1600704.
- 16 Z. Li, *et al.*, Cs₂AgBiBr₆-based heterojunction photodetector for weak-light imaging application, *Surf. Interfaces*, 2022, 29, 101705.
- 17 F. Cao, Z. Li, X. Liu, Z. Shi and X. Fang, Air induced formation of Cs₃Bi₂Br₉/Cs₃BiBr₆ bulk heterojunction and its dual-band photodetection abilities for light communication, *Adv. Funct. Mater.*, 2022, 32(46), 2206151.
- 18 S. Baturay, A. Tombak, D. Batibay and Y. Ocak, n-Type conductivity of CuO thin films by metal doping, *Appl. Surf. Sci.*, 2019, 477, 91–95.
- 19 M. Zhao, Q. Zhao, J. Qiu, H. Xue and H. Pang, Tin-based nanomaterials for electrochemical energy storage, *RSC Adv.*, 2016, 6(98), 95449–95468.
- 20 A. Akbulatov, *et al.*, Comparative intrinsic thermal and photochemical stability of Sn (II) complex halides as next-generation materials for lead-free perovskite solar cells, *J. Phys. Chem. C*, 2019, 123(44), 26862–26869.
- 21 R. Saad, *et al.*, Enhanced CO₂ gas sensing at room temperature using Ag-plated Na-doped CuO thin films synthesized by successive ionic layer adsorption and reaction technique, *Surf. Interfaces*, 2024, 44, 103789.
- 22 N. Planken, W. Kautek and T. Dirmo, Aqueous Spray Pyrolysis of Cu₂O Films: Influence of Reducing Agent and Acetic Acid Addition, *ChemNanoMat*, 2020, 6(4), 663–671.
- 23 M. Dosa, *et al.*, Cerium-Copper Oxides Synthesized in a Multi-Inlet Vortex Reactor as Effective Nanocatalysts for CO and Ethene Oxidation Reactions, *Catalysts*, 2022, 12(4), 364.
- 24 A. Ali, *et al.*, Synthesis of lead-free Cu/CuFeO₂/CZTS thin film as a novel photocatalytic hydrogen generator from wastewater and solar cell applications, *Opt. Quantum Electron.*, 2024, 56(5), 1–22.
- 25 M. Migahed, M. El-Rabiei, H. Nady, H. Goma and E. Zaki, Corrosion inhibition behavior of synthesized imidazolium ionic liquids for carbon steel in deep oil wells formation water, *J. Bio-Tribo-Corros.*, 2017, 3, 1–20.
- 26 G. Xiong, *et al.*, Highly efficient and stable broadband near-infrared-emitting lead-free metal halide double perovskites, *J. Mater. Chem. C*, 2021, 9(38), 13474–13483.
- 27 S. Lim, H. Lin, W. Tsai, H. Lin, Y. Hsu and H. Tuan, Binary halide, ternary perovskite-like, and perovskite-derivative nanostructures: hot injection synthesis and optical and photocatalytic properties, *Nanoscale*, 2017, 9(11), 3747–3751.
- 28 A. Ali, *et al.*, Preparation of inorganic lead-free CuO/Cs₂SnCl₆-KI perovskite for green hydrogen production from wastewater by using solar energy, *J. Photochem. Photobiol., A*, 2023, 445, 115102.
- 29 H. Goma, M. El-Rabiei, H. Nady, E. Zaki and M. Migahed, 1-(2-Aminoethyl)-1-dodecyl-2-undecyl-4, 5-dihydro-1 H-imidazole-1-ium chloride, 1-(2-Aminoethyl)-1-dodecyl-2-tridecyl-4, 5-dihydro-1 H-imidazole-1-ium chloride as Corrosion Inhibitors for Carbon Steel in Oil Wells Formation Water, *Z. für Phys. Chem.*, 2019, 233(11), 1529–1552.
- 30 A. Liu, *et al.*, Room-temperature solution-synthesized p-type copper (I) iodide semiconductors for transparent thin-film transistors and complementary electronics, *Adv. Mater.*, 2018, 30(34), 1802379.
- 31 D. Sarma, C. Malliakas, K. Subrah, S. Islam and M. Kanatzidis, K_{2x}Sn_{4-x}S_{8-x} (x= 0.65–1): a new metal sulfide for rapid and selective removal of Cs⁺, Sr²⁺ and UO₂²⁺ ions, *Chem. Sci.*, 2016, 7(2), 1121–1132.
- 32 Z. Chen, *et al.*, Thin single crystal perovskite solar cells to harvest below-bandgap light absorption, *Nat. Commun.*, 2017, 8(1), 1890.
- 33 A. Ali, *et al.*, Preparation and Characterization of Nanostructured Inorganic Copper Zinc Tin Sulfide-Delafossite Nano/Micro Composite as a Novel Photodetector with High Efficiency, *Photonics*, 2022, 9(12), 979.
- 34 N. Lopez, F. Rodriguez, M. Hernandez, F. Flores and S. Jimenez, Dual-doped CdSe:Cu:O films grown by sputtering using CdSe-CuO composite targets, *J. Phys.:Condens. Matter*, 2020, 32(19), 195701.
- 35 C. Chen, W. Yu, T. Liu, S. Cao and Y. Tsang, Graphene oxide/WS₂/Mg-doped ZnO nanocomposites for solar-light catalytic and anti-bacterial applications, *Sol. Energy Mater. Sol. Cells*, 2017, 160, 43–53.
- 36 D. Ju, X. Jiang, H. Xiao, X. Chen, X. Hu and X. Tao, Narrow band gap and high mobility of lead-free perovskite single crystal Sn-doped MA₃Sb₂I₉, *J. Mater. Chem. A*, 2018, 6(42), 20753–20759.
- 37 A. Nazligul, M. Wang and K. Choy, Recent development in earth-abundant kesterite materials and their applications, *Sustainability*, 2020, 12(12), 5138.
- 38 E. Hamden, *et al.*, Charge-coupled devices detectors with high quantum efficiency at UV wavelengths, *J. Astron. Telesc. Instrum. Syst.*, 2016, 2(3), 036003.
- 39 A. Kumar, M. Khan and M. Kumar, Recent advances in UV photodetectors based on 2D materials: A review, *J. Phys. D:Appl. Phys.*, 2021, 55(13), 133002.
- 40 A. Perveen, *et al.*, Solution processed and highly efficient UV-photodetector based on CsPbBr₃ perovskite-polymer composite film, *J. Photochem. Photobiol., A*, 2022, 426, 113764.
- 41 A. Khan, *et al.*, Advancements in the stability, protection and lead-free strategies of perovskite solar cells: A critical review, *Environ. Sci.:Adv.*, 2024, 3(7), 1004–1029.
- 42 M. Malik, M. Iqbal, J. Choi and P. Pham, 2D materials for efficient photodetection: Overview, mechanisms, performance and UV-IR range applications, *Front. Chem.*, 2022, 10, 905404.
- 43 N. Shinde, P. Deshmukh, S. Patil and C. Lokhande, Development of polyaniline/Cu₂ZnSnS₄ (CZTS) thin film based heterostructure as room temperature LPG sensor, *Sens. Actuators, A*, 2013, 193, 79–86.
- 44 C. Chen, W. Yu, T. Liu, S. Cao and Y. Tsang, Graphene oxide/WS₂/Mg-doped ZnO nanocomposites for solar-light catalytic and anti-bacterial applications, *Sol. Energy Mater.*, 2017, 160, 43–53.



- 45 X. Wang, W. Tian, M. Liao, Y. Bando and D. Golberg, Recent advances in solution-processed inorganic nanofilm photodetectors, *Chem. Soc. Rev.*, 2014, **43**(5), 1400–1422.
- 46 L. Luo, *et al.*, Light trapping and surface plasmon enhanced high-performance NIR photodetector, *Sci. Rep.*, 2014, **4**(1), 1–8.
- 47 H. Zhao, B. Ouyang, L. Han, Y. Mishra, Z. Zhang and Y. Yang, Conjoined photo-thermoelectric effect in ZnO–graphene nanocomposite foam for self-powered simultaneous temperature and light sensing, *Sci. Rep.*, 2020, **10**(1), 1–11.
- 48 S. Kunwar, S. Pandit, J. Jeong and J. Lee, Improved photoresponse of UV photodetectors by the incorporation of plasmonic nanoparticles on GaN through the resonant coupling of localized surface plasmon resonance, *Nano-Micro Lett.*, 2020, **12**(1), 1–16.
- 49 R. Jia, D. Zhao, N. Gao and D. Liu, Polarization enhanced charge transfer: dual-band GaN-based plasmonic photodetector, *Sci. Rep.*, 2017, **7**(1), 1–8.
- 50 S. Bell, G. Will and J. Bell, Light intensity effects on photocatalytic water splitting with a titania catalyst, *Int. J. Hydrogen Energy*, 2013, **38**(17), 6938–6947.
- 51 M. El-Rabiei, G. AbdElHafez and A. Ali, Effects of Alloying Elements (Ti and x Al) on the Electrochemical Corrosion Behaviour of Iron-Based Alloys in Corrosive Solutions of Different pH, *J. Bio-Tribo-Corros.*, 2020, **6**, 1–16.
- 52 M. Shaban, M. Rabia, A. Abd ElSayed, A. Ahmed and S. Sayed, Photocatalytic properties of PbS/graphene oxide/polyaniline electrode for hydrogen generation, *Sci. Rep.*, 2017, **7**(1), 1–13.
- 53 M. Fadel, M. Rabia, S. Ezzat, N. Mansour, E. Saeed and S. Sayyah, Effect of annealing temperature on VO₂(M)/ITO film nanomaterials for thermochromic smart windows application and study its contact angle, *Nanophotonics*, 2018, **12**(1), 016009.
- 54 N. Huo and G. Konstantatos, Recent progress and future prospects of 2D-based photodetectors, *Adv. Mater.*, 2018, **30**(51), 1801164.
- 55 P. Martyniuk, J. Antoszewski, M. Martyniuk, L. Faraone and A. Rogalski, New concepts in infrared photodetector designs, *Appl. Phys. Rev.*, 2014, **1**(4), 041102.
- 56 D. Guo, W. Li, D. Wang, B. Meng, D. Fang and Z. Wei, High performance Cu₂O film/ZnO nanowires self-powered photodetector by electrochemical deposition, *Chin. Phys. B*, 2020, **29**(9), 098504.
- 57 H. Madusanka, H. Herath and C. Fernando, High photoresponse performance of self-powered n-Cu₂O/p-CuI heterojunction based UV-Visible photodetector, *Sens. Actuators, A*, 2019, **296**, 61–69.
- 58 F. Cao, W. Tian, M. Wang, M. Wang and L. Li, Stability enhancement of lead-free CsSnI₃ perovskite photodetector with reductive ascorbic acid additive, *InfoMat*, 2020, **2**(3), 577–584.
- 59 M. Han, *et al.*, Controllable growth of lead-free all-inorganic perovskite nanowire array with fast and stable near-infrared photodetection, *J. Phys. Chem. C*, 2019, **123**(28), 17566–17573.
- 60 T. Lan, A. Fallatah, E. Suiter and S. Padalkar, Size controlled copper (I) oxide nanoparticles influence sensitivity of glucose biosensor, *Sens. Actuators, A*, 2017, **17**(9), 1944.
- 61 X. Han, *et al.*, Lead-free double perovskite Cs₂SnX₆: facile solution synthesis and excellent stability, *Small*, 2019, **15**(39), 1901650.
- 62 M. Rabia, M. Shaban, B. Jibali and A. Abdelkhaliek, Effect of annealing temperature on the photoactivity of ITO/VO₂(M)/Au film electrodes for water splitting, *Nanosci. Nanotechnol.*, 2020, **20**(7), 4120–4130.
- 63 F. Alkallas, A. Ben, T. Alrebdi, A. Ahmed and M. Rabia, Development of a Highly Efficient Optoelectronic Device Based on CuFeO₂/CuO/Cu Composite Nanomaterials, *Materials*, 2022, **15**(19), 6857.
- 64 A. Abdelazeez, *et al.*, Development of CuO nanoporous material as a highly efficient optoelectronic device, *Appl. Phys. A: Mater. Sci. Process.*, 2022, **128**(4), 321.
- 65 L. Su, W. Yang, J. Cai, H. Chen and X. Fang, Self-powered ultraviolet photodetectors driven by built-in electric field, *Small*, 2017, **13**(45), 1701687.
- 66 M. Zayed, S. Samy, M. Shaban, A. Altowyan, H. Hamdy and A. Ahmed, Fabrication of TiO₂/NiO pn nanocomposite for enhancement dye photodegradation under solar radiation, *Nanomaterials*, 2022, **12**(6), 989.
- 67 T. Bu, *et al.*, A novel quadruple-cation absorber for universal hysteresis elimination for high efficiency and stable perovskite solar cells, *Energy Environ. Sci.*, 2017, **10**(12), 2509–2515.
- 68 Z. Tang, *et al.*, Hysteresis-free perovskite solar cells made of potassium-doped organometal halide perovskite, *Sci. Rep.*, 2017, **7**(1), 12183.

