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Construction of a Rh-doped $\text{SrTiO}_3/\text{g-C}_3\text{N}_4$ p–n heterojunction for enhanced photoelectrochemical performance†

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Semiconductor photoelectrochemical (PEC) technology, capable of generating hydrogen energy from water and solar energy, has emerged as a promising solution to address environmental and energy challenges. Rhodium-doped strontium titanate (Rh doped SrTiO_3 , RST) is regarded as a highly promising photocathode material for PEC systems. Nevertheless, the PEC performance of RST is hindered by inefficient carrier separation and poor charge transfer properties. In this paper, a hybrid RST/ $\text{g-C}_3\text{N}_4$ heterojunction sample was prepared by a simple method. The pure RST sample possesses the ability to absorb visible light and exhibit p-type semiconductor characteristics with a negative photocurrent value of $-6.2 \mu\text{A cm}^{-2}$, and can be used as a photocathode material in PEC systems. The enhanced PEC performance with larger photocurrent values can be observed in the heterojunction samples. A photocurrent value of $-26.5 \mu\text{A cm}^{-2}$ was achieved in the optimized heterojunction, which is 4.3 times higher than the RST sample. This enhanced PEC performance can be attributed to the formation of a p–n junction between RST and $\text{g-C}_3\text{N}_4$ in composite samples. The p–n junction with built-in electric field and good band alignment will facilitate the more efficient separation and transfer of carriers, leading to enhanced PEC performance.

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1. Introduction

In the last few decades, semiconductor photoelectrochemical (PEC) technology has attracted great attention due to its production of hydrogen (H_2) from water and solar energy.^{1–4} Recently, the development of photoelectrodes that can absorb visible light, which accounts for approximately 45% of the total solar energy, has been the research focus. Several semiconductor photoelectrodes, including WO_3 , Fe_2O_3 , and BiVO_4 , have been developed for PEC water splitting, leveraging their visible-light absorption capabilities.^{5–7} These oxide photoelectrodes exhibit an inadequate conduction band level for H_2 evolution with an n-type semiconductor character. Therefore, an external bias is necessary for water splitting in PEC systems utilizing these photoelectrodes. Rhodium-doped strontium titanate (Rh doped SrTiO_3 , RST) photoelectrode materials

exhibit the ability to absorb visible light, with their conduction band levels exceeding the redox potential required for H_2 evolution. Consequently, they are suitable as photocathodes for hydrogen production in PEC systems, attributed to their p-type semiconducting properties.^{8–12} The visible light absorption ability of RST can be attributed to electronic transitions from the Rh^{3+} (donor levels) to the conduction band (Ti 3d orbitals) and the electronic transitions from valence band (O 2p) to Rh^{4+} (acceptor levels).^{13–16} However, the photoelectric conversion efficiency is still limited due to the high recombination rates and poor transfer properties of carriers in RST. To address the issues, coupling RST with other materials to form a heterostructure has emerged as an efficient strategy.^{17–21}

Recently, n-type graphitic carbon nitride ($\text{g-C}_3\text{N}_4$), known for its thermal and chemical stability, has emerged as a promising material in the field of photocatalysis. As an excellent visible light photocatalyst, $\text{g-C}_3\text{N}_4$ possesses a suitable bandgap of 2.7 eV, with band potentials well-suited for water splitting, making it an ideal candidate for renewable energy generation.^{22,23} However, pure $\text{g-C}_3\text{N}_4$ faces challenges in separating photogenerated charges and exhibits low electrical conductivity, resulting in suboptimal photocatalytic performance. Constructing a p–n heterojunction can utilize the internal electric field to prevent the rapid recombination of photo-generated carriers, thereby enhancing photocatalytic activity. Various semiconductor/ $\text{g-C}_3\text{N}_4$ p–n heterojunctions (such as

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NiO/g-C₃N₄, CoFe₂O₄/g-C₃N₄, Co₃O₄/g-C₃N₄ and Mn₃O₄/g-C₃N₄) have been developed to reduce carrier recombination rates.^{24–27} In the past decade, RST and g-C₃N₄ composites have been extensively utilized in photocatalysis. For instance, Wang *et al.* prepared the RST and phosphorus doped g-C₃N₄ composites by calcination treatment.²⁸ The results possessed the rational band alignment can greatly promote the charge separation and thus leading to the superior activity for H₂ production in composites. Kumar *et al.* fabricated the 3D superstructure of g-C₃N₄/RGO/RST for H₂ production.²⁹ The higher catalytic activity can be attributed to the well-matched energy band positions, facilitating efficient charge transfer in composite photocatalysts. However, there was lacking research on the dynamics and PEC performance of p–n junction photoelectrodes by combining p-type RST to n-type g-C₃N₄.

In this work, we design and fabricate RST and g-C₃N₄ composite photoelectrodes using a straightforward scraping coating technique. The scraping coating method presents potential advantages for large-area electrode fabrication, including simplicity, low cost, and broad applicability. The experimental data demonstrates that incorporating g-C₃N₄ materials forms p–n junctions in the RST and g-C₃N₄ composites, significantly improving the PEC properties. The pure RST photoelectrode sample has a photocurrent value of $-6.2 \mu\text{A cm}^{-2}$ with the p-type characters in PEC measurement. The maximal value of $-26.5 \mu\text{A cm}^{-2}$ exhibited in the RSTCN7.5 sample, which was 4.3 times than that of RST sample. The enhanced PEC performance can be attributed to the formed p–n junction between RST and g-C₃N₄, which will improve the dynamics of carrier separation and transport, leading to larger photocurrent values in heterojunction photoelectrodes. Our experimental results provide a facile and effective method for integrating p–n heterojunction materials into PEC applications.

2. Materials and methods

2.1 Chemical and materials

SrCO₃ (99.9%), TiO₂ (99.9%), and melamine (99%) were purchased from Sigma Aldrich. Rh₂O₃ (99.9%) was purchased from J&K Scientific. Na₂SO₄ (99%) and ethanol (99.7%) were obtained from Sinopharm chemical reagent Co. Ltd. All chemicals were used for the experiments without further purification. Fluorine-doped SnO₂ (FTO) glass substrates were purchased from South China Xiang Cheng Technology Co. Ltd, and deionized water can be obtained from local sources.

2.2 Preparation of photoelectrodes

In this study, SrTiO₃ doped with Rh was prepared by the solid-state reaction.¹⁰ Considering the influence of Rh doping on the onset potential of the RST sample, we chose 2.5% Rh doping.¹⁶ The starting materials, SrCO₃ (1.5796 g), TiO₂ (0.7790), and Rh₂O₃ (0.0317 g), were mixed at an agate mortar and the chemical formula was Sr(Ti_{0.975}Rh_{0.025})O₃. Then, the mixtures were calcined at 900 °C for 1 h and at 1100 °C for 10 h in air. The obtained powders are labeled as RST. The pure SrTiO₃ powder was prepared by the same method without Rh doping. The g-

C₃N₄ was synthesized *via* the thermal polymerization method by heating melamine at 550 °C for 4 h in air. The prepared RST and g-C₃N₄ powders were mixed with 50 mL of deionized water and stirred continuously for 10 hours to achieve homogeneous suspensions. The suspensions were dried at 80 °C and then calcined at 450 °C for 2 hours in air. During this process, the weight percentages of g-C₃N₄ in the RSTCN composites were controlled at 2.5%, 5%, 7.5%, and 10%. Accordingly, the resulting samples were named RSTCN2.5, RSTCN5, RSTCN7.5, and RSTCN10.

The FTO substrates were sequentially cleaned by ultrasonic cleaning with acetone, ethanol, and deionized water, respectively, and then the FTO substrates were dried for further use. The photoelectrodes were prepared by the simple knife scraping method. Briefly, 50 mg RSTCN composites powder and 100 μL ethanol solution for grinding to obtain the slurry. Then the slurry materials were scraped on the FTO substrate with the annealing treatment at 450 °C for 2 hours in air. Pure RST photoelectrodes were also synthesized using the same method. The annealing process will have a positive effect on the adhesion between the FTO substrates and photoelectrode materials, also will facilitate the transfer of charge carriers in sample. Pure g-C₃N₄ photoelectrodes were prepared using the scrape method. A mixture of 50 mg of g-C₃N₄ powder and 100 μL of ethanol containing 0.5 wt% Nafion was ground in an agate mortar to form a viscous slurry. This slurry was applied to FTO substrates and calcined at 200 °C for 2 hours in air. The addition of Nafion and low-temperature annealing helps prevent the detachment of g-C₃N₄.

2.3 Characterization methods

The morphology structures were investigated by field emission scanning electron microscope (FESEM, Hitachi S8100) and transmission electron microscope (TEM, Tecnai G2 F20). Fourier transform infrared (FTIR) spectra were measured on a Varian 3600 FTIR spectrometer. The X-ray diffraction (XRD) patterns were measured by Bruker D8 Advance X-ray diffractometer equipped with Cu K α radiation ($\lambda = 1.540598 \text{ \AA}$) to analyze the crystal structures of different samples. The X-ray photoelectron spectrometer (XPS, PerkinElmer PHI 5000) curves were measured to determine the chemical states and valence band spectra for different samples. And the C 1s peak of 284.80 eV served as a reference for calibration and correction. The optical absorption spectra of different samples were obtained by the UV/Vis/NIR Spectrometer (PerkinElmer Lambda 750). The photoluminescence (PL) spectra were conducted by the steady fluorescence spectrometer (FLS1000, Edinburgh) at the excitation wavelength of 320 nm.

2.4 Photoelectrochemical measurement

The CHI-660E electrochemical workstation was used to evaluate the PEC performance on different samples in a standard three-electrode system. The photoelectrodes served as the working electrode, while the Pt wire and Ag/AgCl electrode functioned as the counter and reference electrodes, respectively. The 0.5 mol L⁻¹ aqueous solution of Na₂SO₄ was chosen as the



electrolyte. The light source is a 300 W Xe lamp, and the light intensity was controlled at 100 mW cm^{-2} . Electrochemical impedance spectroscopy (EIS) was measured within the frequency range of 0.01 Hz to 10 kHz, applying a 10 mV AC voltage. The Mott-Schottky plots were carried out at a fixed frequency of 1000 Hz.

3. Results and discussion

3.1 Morphology and structure characterization

Fig. 1a displays the morphology of pure RST with an irregular particle shape. After decorating with $\text{g-C}_3\text{N}_4$ materials, there are some subtle but obvious changes in RSTCN7.5 sample. The curved materials were the existence of $\text{g-C}_3\text{N}_4$ in RSTCN7.5 sample. As shown in Fig. 1c, the XRD patterns were carried out to observe the crystal structure of different samples. For the RST sample, diffraction peaks corresponding to the crystal planes (100), (110), (111), (200), (210), (211), (220), and (310) were observed, with their positions and relative intensities consistent with the JCPDS No. 35-0634.¹⁶ However, the diffraction peaks of $\text{g-C}_3\text{N}_4$ were not observed in the RSTCN7.5 sample, which may be caused by the small amount of $\text{g-C}_3\text{N}_4$ in composites. The Fig. 1d shows the FTIR spectra of different samples. For the RST, the peak of Ti-O bond can be observed at 600 cm^{-1} . For the RSTCN7.5 sample, in addition to the peak of Ti-O bond, there are other peaks attributed to $\text{g-C}_3\text{N}_4$. The peaks located at 810 cm^{-1} and $1200\text{--}1600 \text{ cm}^{-1}$ can be assigned to the breathing vibration of triazine units and the C-N/C=N bonds,

respectively.²⁸ The morphology and FTIR spectra for $\text{g-C}_3\text{N}_4$ are displayed in Fig. S1.† Based on above analysis, we have successfully prepared the RSTCN heterojunction samples.

3.2 Chemical states analysis

As shown in Fig. 2, the chemical states of the elements for RST and RSTCN7.5 samples were measured by the XPS spectra. For pure RST sample, the peaks located at binding energy of 309.76 eV, 314.54 eV, 132.90 eV, 134.55 eV, 458.79 eV, and 464.61 eV were observed, which can be assigned to the Rh 3d_{5/2}, Rh 3d_{3/2}, Sr 3d_{5/2}, Sr 3d_{3/2}, Ti 2p_{3/2}, and Ti 2p_{1/2}, respectively.³⁰ After decorating with $\text{g-C}_3\text{N}_4$ materials, a shift towards lower binding energy can be observed on the Rh 3d, Sr 3d, and Ti 2p in RSTCN composites. The negative shift towards lower binding energy can be attributed to the electronic interaction between RST and $\text{g-C}_3\text{N}_4$, and the RST materials obtains electrons from the $\text{g-C}_3\text{N}_4$.^{31,32} For RSTCN7.5 heterojunction sample, the deconvoluted high-resolution C 1s spectra have three peaks at binding energy positions of 284.80, 285.70, and 288.12 eV, which can be assigned to C=C, C-C/C-N and N-C=N, respectively.^{33,34} In addition, the high-resolution N 1s spectra can be deconvoluted into three peaks at binding energy positions of 398.61 eV (C=N-C), 399.49 eV (N-(C)₃), and 401.23 eV (H-N-C), respectively.²⁴ Based on the XPS analysis, the prepared heterojunction samples are not a simple mixture of two components, and the XPS analysis also confirmed the successful preparation of RST/ $\text{g-C}_3\text{N}_4$ heterostructures.

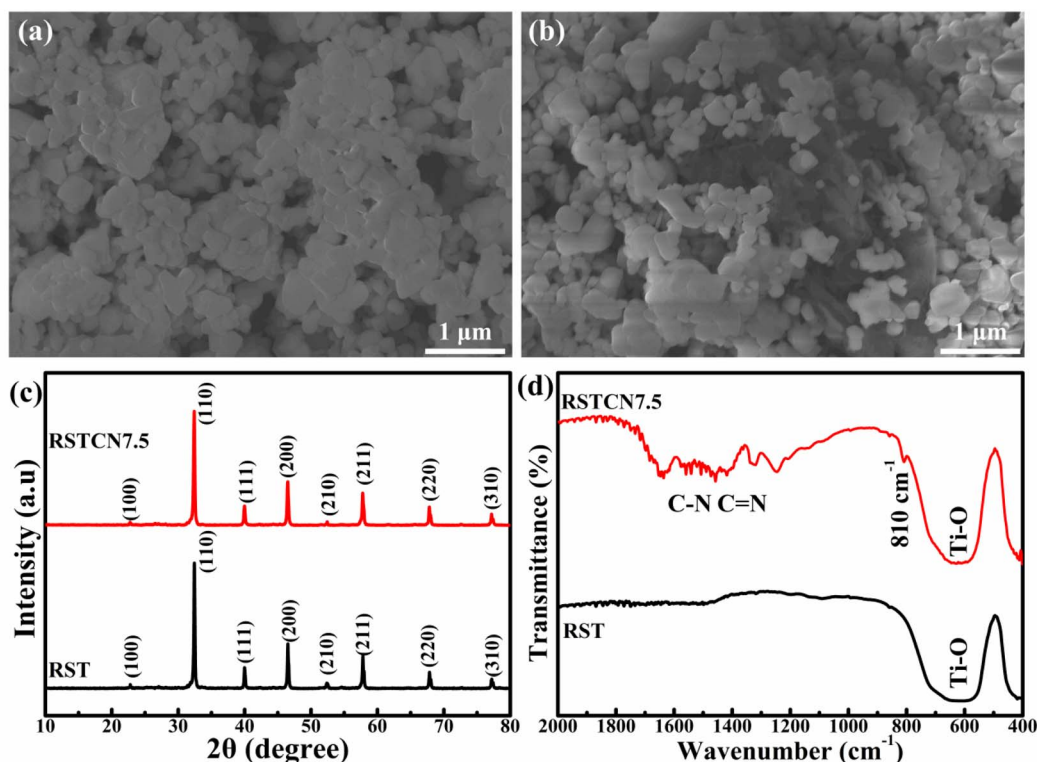


Fig. 1 Basic structure characterisation of different samples. SEM images for (a) pure RST and (b) RSTCN7.5 sample. (c) XRD patterns of RST and RSTCN7.5 samples. (d) FTIR spectra of RST and RSTCN7.5 samples.

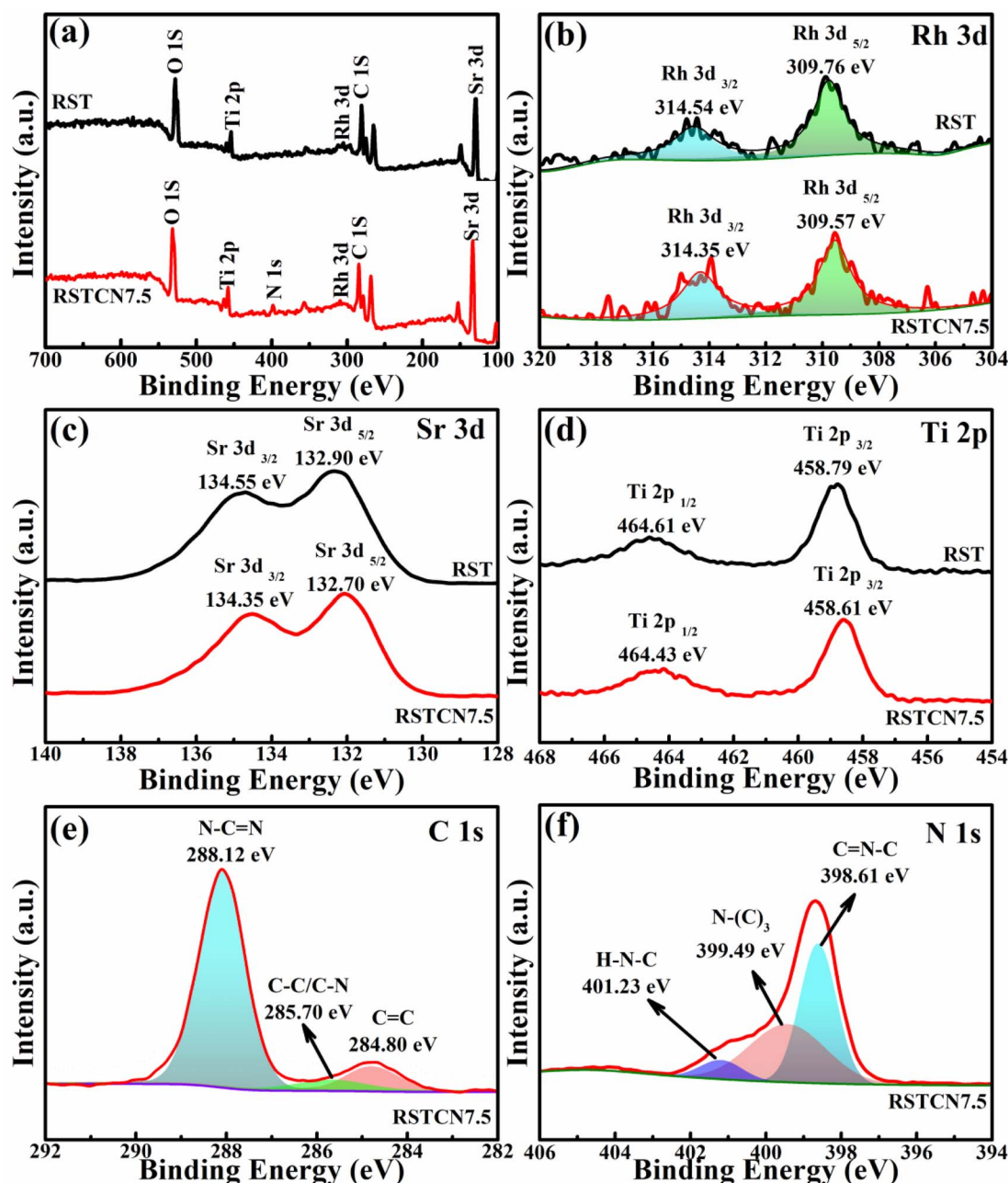


Fig. 2 XPS analysis of different samples. (a) XPS survey spectra, high resolution XPS spectra of (b) Rh 3d, (c) Sr 3d, (d) Ti 2p, (e) C 1s, and (f) N 1s.

3.3 Photoelectrochemical measurements

Fig. 3a–e display the measured current density *versus* time curves of different samples under chopped light irradiation, respectively. All the samples exhibit the clear photoelectric response. A negative photocurrent value of about $-6.2 \mu\text{A cm}^{-2}$ (at 0 V *versus* Ag/AgCl) was investigated in pure RST sample, implying its p-type semiconductor characteristic. In the PEC measurement (Fig. S3†), pure g-C₃N₄ exhibited a lower negative photocurrent value. Consequently, the semiconductor characteristics of various materials will be further verified using Mott-Schottky curves and valence band spectra. After decorating with g-C₃N₄, all the composite heterojunction photoelectrodes show the enhanced PEC properties with the larger photocurrent

values. By increasing the contents of g-C₃N₄ materials, the RSTCN heterojunction samples show an improved photocurrent density and a maximal value of $-26.5 \mu\text{A cm}^{-2}$ (at 0 V *versus* Ag/AgCl) in the optimized RSTCN7.5 heterojunction sample, which is 4.3 times than that in pure RST. However, the reduction of photocurrent density value will be investigated with the further increase of g-C₃N₄ materials content. It is important to note that the simplicity and low cost of the scraping method used for powder photoelectrodes in this study resulted in photocurrent values as low as microamperes. The PEC properties were inferior to those of other photocathode samples, such as Si and Cu₂O photocathodes, as well as the high-quality RST thin film photoelectrode prepared by the molecular beam



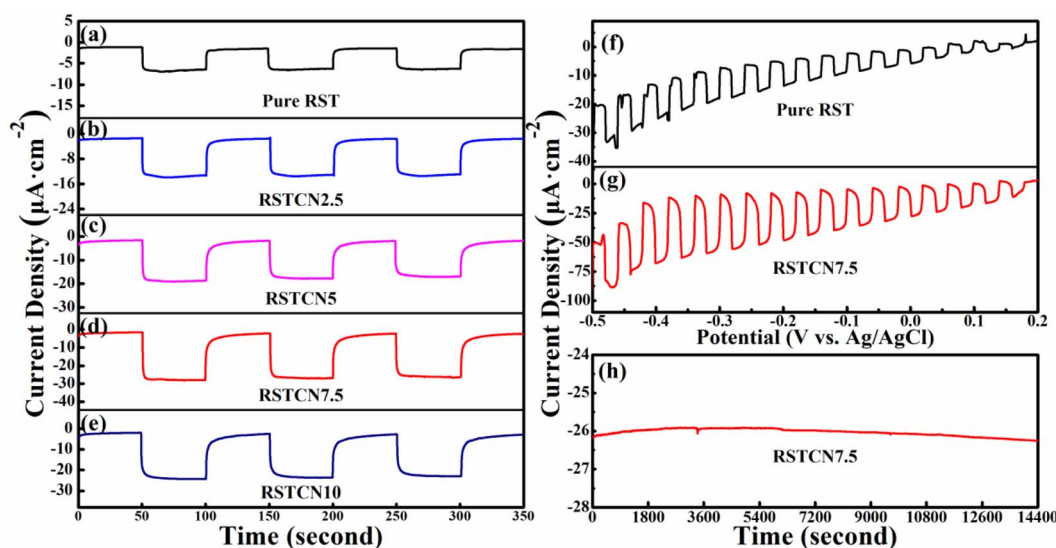


Fig. 3 PEC properties of different samples. Photocurrent versus time under chopped light (a) pure RST, (b) RSTCN2.5, (c) RSTCN5, (d) RSTCN7.5 and (e) RSTCN10. LSV curves under chopped light (f) pure RST and (g) RSTCN7.5 sample. (h) Longtime photocurrent versus time under light for 4 hours of RSTCN7.5 sample.

epitaxy (MBE) method.^{11,35,36} To further observe the performance of the samples, the linear sweep voltammetry (LSV) curves of pure RST and RSTCN7.5 heterojunction samples were obtained under chopped light irradiation (Fig. 3f and g). The photocurrent density values have a steady increase with the increasing applied potential, and the good photocurrent responses are observed for each switch on and switch off in both RST and RSTCN7.5 heterojunction samples. Compared to pure RST sample, RSTCN7.5 heterojunction sample exhibited stronger response, indicating that the superior utilization of the light energy conversion. In addition, Fig. 3h shows the photocurrent stability test of the RSTCN7.5 sample in 0.5 M Na_2SO_4 solution under illumination for 4 hours, indicating the high stability of the PEC system.

3.4 Optical characterization

Generally, the improved PEC properties in semiconductor heterostructures can be attributed primarily to two factors. One factor is the enhanced light absorption capacity, whereas the other is the higher separation efficiency and transport properties of carrier.^{37,38} Fig. 4a and S4a† demonstrates the absorption spectra of different samples. The absorption edge of pure SrTiO_3 was observed to be approximately 380 nm, located within the ultraviolet (UV) region. By doping of Rh element, the light absorption will be extended to the visible region. However, after incorporating $\text{g-C}_3\text{N}_4$, the optical absorption performance of the composite samples decreases, and the light absorption capacity of the composites diminishes with increasing $\text{g-C}_3\text{N}_4$ content. To gain more insights from the UV-vis absorption spectra, we can estimate the band gap energies of various samples by extrapolating the linear region of the $(\alpha h\nu)^2$ versus $(h\nu)$ plot to zero. Fig. S4b† displays that the band gap value of 3.26 eV in pure SrTiO_3 (Fig. S4b†), while the values of 1.65 and 2.65 eV can

be obtained in RST sample (Fig. 4b). On referring to results and previous literature, the calculated values of 2.65 eV can be attributed to the electronic transitions from Rh^{3+} (electron donor levels) to conduction band of SrTiO_3 , while that 1.65 eV can be attributed to the electronic transitions from valence band of SrTiO_3 to Rh^{4+} (electron acceptor levels).^{13,14,16} For pure $\text{g-C}_3\text{N}_4$, the band gap value of $\text{g-C}_3\text{N}_4$ can be estimated to be 2.76 eV (Fig. 4c), that means the ability of light absorption in composite sample cannot be enhanced by decorating with $\text{g-C}_3\text{N}_4$ in composite sample, which is consistent with the absorption spectra. That is to say, it can be ruled out that the enhanced light absorption ability leads to the improvement of PEC performance in RSTCN heterojunctions.

3.5 PL analysis

To conduct a comprehensive assessment of the charge carrier separation and recombination process in different samples, the photoluminescence (PL) spectra were conducted. In general, PL emission primarily arises from the recombination of photo-generated electrons and holes in semiconductors, and a lower PL intensity indicates a reduced recombination rate of electron-hole pairs.³⁹ As shown in Fig. 4c, an emission peak at 450 nm, attributed to radiative carrier recombinations, can be observed in the pure RST sample. Compared to the RST sample, all RSTCN composites showed reduced PL intensity, indicating decreased carrier recombination rates. As the $\text{g-C}_3\text{N}_4$ content in the composites increases, the PL intensity initially decreases and then increases, with the RSTCN7.5 heterojunction sample showing the lowest PL peak intensity. The RSTCN10 sample exhibits a higher PL peak intensity compared to the RSTCN7.5 sample, indicating higher recombination rates, which lead to decreased PEC properties in the RSTCN10 sample. To further investigate the carrier separation and recombination processes

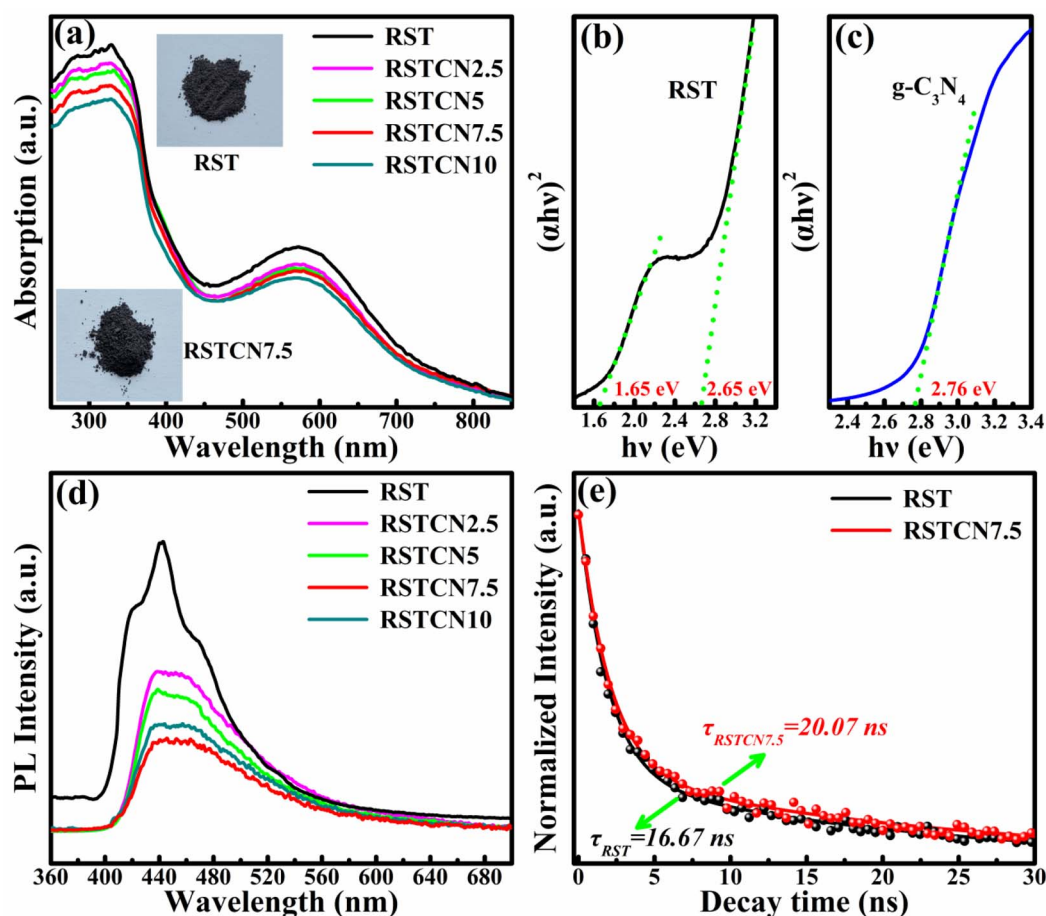


Fig. 4 Optical properties and carrier kinetics of different samples. (a) Absorption spectra of different samples. E_g values for (b) pure RST, (c) $g\text{-C}_3\text{N}_4$. (d) PL spectra of different samples, (e) PL-decay of RST and RSTCN7.5.

in the samples, PL decay curves for RST and RSTCN7.5 were recorded at a detection wavelength of 450 nm. The curves were fitted using the double exponential function $I(t) = A_1 e^{-t/\tau_1} + A_2 e^{-t/\tau_2} + I(0)$. The carrier lifetime can be determined by the $\tau = (A_1\tau_1^2 + A_2\tau_2^2)/(A_1\tau_1 + A_2\tau_2)$.⁴⁰ As shown in Fig. 5d and Table S1,[†] the carrier lifetimes for RST and RSTCN7.5 are 16.67 ns and 20.07 ns, respectively. Additionally, the PL peak intensity of pure $g\text{-C}_3\text{N}_4$ is higher than that of RST, while its carrier lifetime is shorter (Fig. S5 and Table S1[†]). Clearly, the RSTCN7.5 heterojunction sample exhibits the longest carrier lifetime, indicating superior separation efficiency and transmission performance of carriers within the composite material.

3.6 EIS analysis

To evaluate the kinetics of charge transfer process, electrochemical impedance spectroscopy (EIS) was conducted on various samples without light irradiation. Generally, the arc radius obtained from EIS serves as an indicator of the charge transfer resistance occurring at the interface between the semiconductor materials and the electrolyte.^{41,42} As depicted in Fig. 5a, all samples exhibit only one arc, suggesting a significant resistance between the semiconductor photoelectrodes and electrolyte. In addition, the equivalent circuit (EC, inset of Fig. 5a) model was employed to fit the EIS data to enhance the

analysis of EIS results. The fitting results are summarized in Table S2,[†] indicating that the charge transfer resistance (R_{ct}) at the semiconductor/electrolyte interface for the RSTCN7.5 heterojunction sample (3699 Ω) is significantly lower than that of the pure RST (6769 Ω). The reduced R_{ct} suggests a more efficient transfer of carriers across the interface to the electrolyte, thereby suppressing carrier recombination, which is consistent with the PL analysis. Moreover, the lower slope of Mott-Schottky (MS) curves in RSTCN7.5 sample also indicates that the higher charge carrier densities in heterojunction sample. Based on the results, the extended light absorption capacity leading to enhanced PEC properties of RSTCN heterojunction can be excluded. Instead, the enhanced PEC performance observed in RSTCN heterojunction can be attributed to the reduced recombination rates, improved carrier higher separation efficiency, and enhanced carrier transport properties.

3.7 Mott-Schottky and band edge positions

As shown in Fig. 5, the Mott-Schottky (MS) curves were employed to assess the semiconductor characteristics of different samples. The negative slope of linear segment for RST sample is investigated, which indicates its p-type semiconductor characteristics (Fig. 5b). The positive slope of linear segment for $g\text{-C}_3\text{N}_4$ material indicates its n-type characteristics



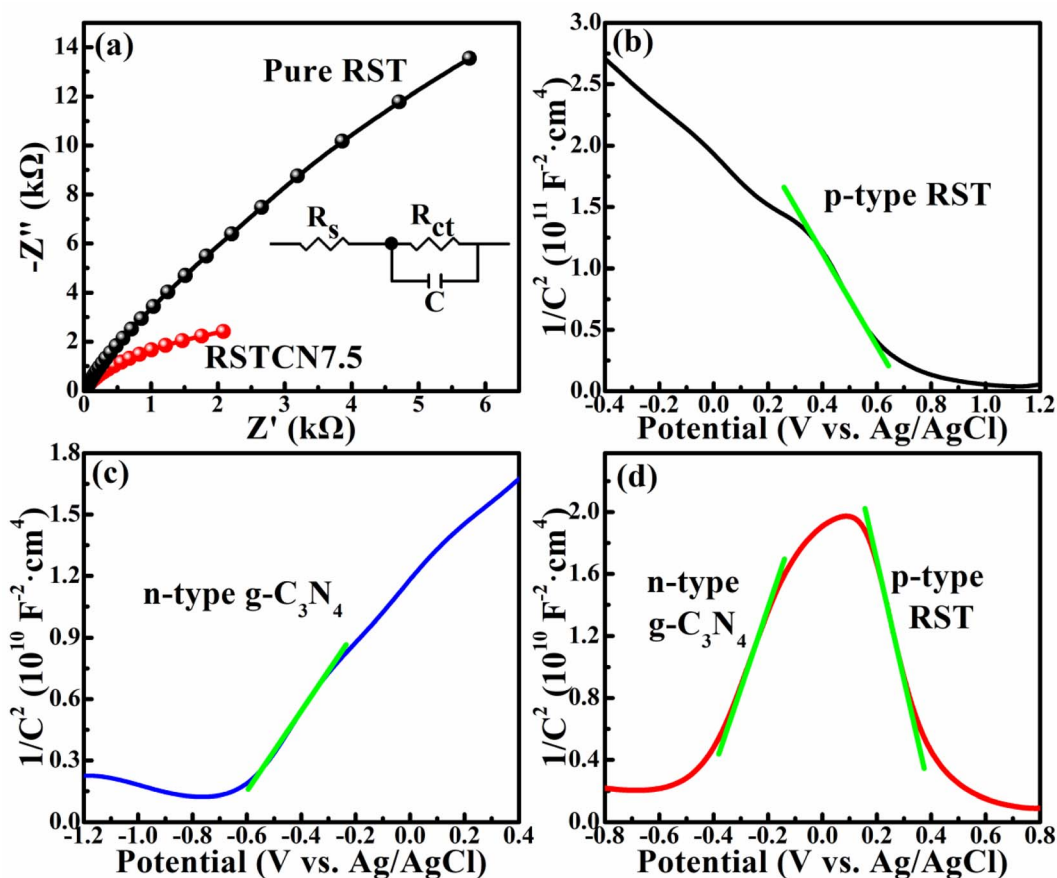


Fig. 5 (a) EIS of RST and RSTCN7.5 samples. Mott-Schottky curves of different samples. (b) RST, (c) g-C₃N₄, and (d) RSTCN7.5 heterojunction.

(Fig. 5c). However, the obtained MS curve exhibited an inverted “V” shape with two linear segments, indicating the existence of p-n junction in composite sample.^{43,44} The linear segment with a negative and positive slope can be attributed to p-type RST and n-type g-C₃N₄ materials (Fig. 5d), respectively. In this study, the band edge positions at the point of zero charge can be determined by the following equations:^{45,46}

$$E_{VB} = \chi - 4.5 \text{ eV} + 0.5 \times E_g \quad (1)$$

$$E_{CB} = E_{VB} - E_g \quad (2)$$

where the E_{CB} , E_{VB} , χ and E_g represent conduction band, valence band, electronegativity value, and the gap energies of semiconductors, respectively. The electronegativity value of SrTiO₃ is

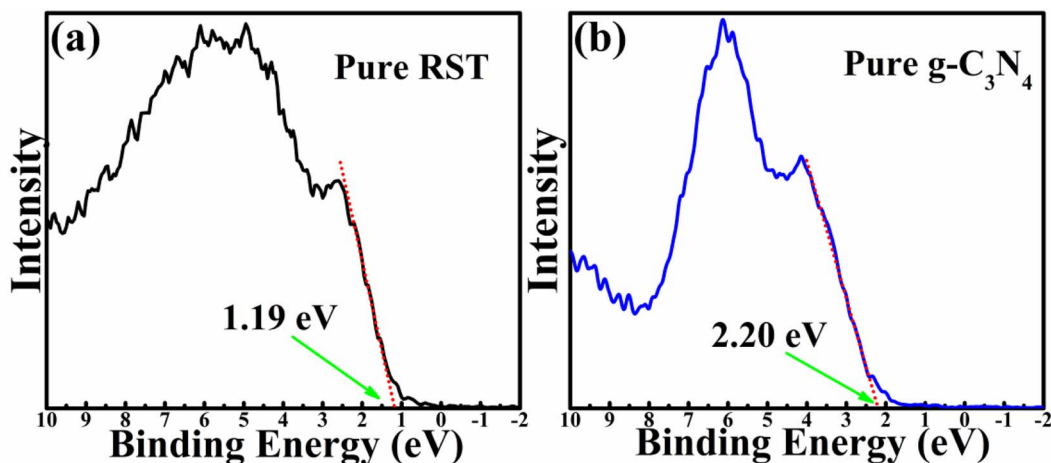


Fig. 6 XPS valence band spectra of different samples. (a) RST and (b) g-C₃N₄.



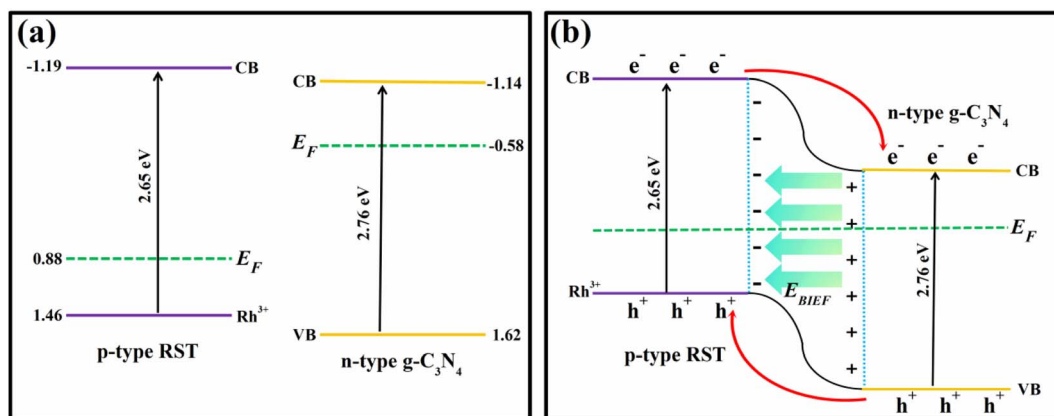


Fig. 7 Energy diagram for enhanced PEC performance. (a) Band edges position between RST and g-C₃N₄ before contact, (b) band alignment between RST and g-C₃N₄ after contact (E_{BIEF} : built-in electric field).

4.94 eV, while that of g-C₃N₄ is 4.74 eV, respectively.^{45,46} According to empirical equations, the band positions of E_{CB} and E_{VB} in SrTiO₃ are -1.19 eV and 2.07 eV, respectively, while that in g-C₃N₄ are -1.14 eV and 1.62 eV, respectively. After doping Rh element, the band positions of Rh⁴⁺ and Rh³⁺ in RST are 1.46 eV and 0.42 eV, respectively. The relative difference between Fermi level (E_F) and E_{VB} ($E_F - E_{VB}$) of the semiconductors can be determined by extrapolating the linear region of XPS valence band spectra to zero. As shown in Fig. 6, the $E_F - E_{VB}$ value of RST is 1.19 eV, while the corresponding value of g-C₃N₄ is 2.20 eV.^{24,30} In addition, the $E_F - E_{VB}$ value of RST is less than half of the bandgap due to its p-type semiconductor characteristic, whereas that of g-C₃N₄ exceeds half of the bandgap due to its n-type characteristic.

Based on the above analysis, the positions of pure RST, including Rh⁴⁺ and Rh³⁺ levels, are shown in Fig. S7.† The Rh⁴⁺ electron acceptor levels have insufficient potential for H₂ evolution, indicating that they do not significantly contribute to this process, consistent with previous studies.¹⁰ Conversely, Rh³⁺ levels can serve as electron donors, transferring electrons to the E_{CB} of RST for H₂ evolution. Thus, the band position diagram of pure RST (before contact with g-C₃N₄) can be simplified, as depicted in Fig. 7a. The positions for g-C₃N₄ (before contact with RST) are also illustrated in Fig. 7a. After contact, electrons will flow from g-C₃N₄ to RST until the E_F levels reach equilibrium. After the equilibrium, the p-type RST and n-type g-C₃N₄ regions will have negative and positive charges, respectively, resulting in the formation of a built-in electric field (E_{BIEF}), oriented from g-C₃N₄ to RST with the well-matched energy band alignment.⁴⁷ Fig. 7b shows the energy band diagrams of the p-n junction between RST and g-C₃N₄. The E_{CB} of RST is more positive than that of the g-C₃N₄, whereas the E_{VB} of g-C₃N₄ is more negative than the Rh³⁺ of RST. Under light irradiation, the photogenerated electrons will migrate from the E_{CB} of RST to the E_{CB} of g-C₃N₄, whereas the photogenerated holes will migrate in the opposite direction, from the E_{VB} of g-C₃N₄ to the Rh³⁺ of RST. Such charge transfer process induced by p-n junction can greatly reduce the recombination rates and improve the transfer properties of carriers, leading to the enhanced PEC performance in RSTCN heterojunction sample.

4. Conclusions

In conclusion, a hybrid RSTCN p-n heterojunction was prepared by a simple method. The prepared RST photoelectrode demonstrates excellent visible light absorption capabilities and exhibits p-type semiconductor properties, yielding a photocurrent value of $-6.2 \mu\text{A cm}^{-2}$ and demonstrating photocathodic behavior within the PEC system. After decorating with n-type g-C₃N₄ materials, all composite photoelectrodes demonstrated significantly enhanced PEC performance, characterized by increased photocurrent values. The optimized RSTCN7.5 heterojunction sample demonstrates a maximum photocurrent density of approximately $-26.5 \mu\text{A cm}^{-2}$, which is 4.3 times higher than that of the pure RST photoelectrode sample. The PL and EIS analyses revealed the presence of reduced recombination rates and enhanced carrier transfer properties in the heterojunction sample, which can be attributed to the efficient charge separation and transfer process caused by the formation of a p-n junction between RST and g-C₃N₄. The formed p-n junction, featuring a built-in electric field and well-matched energy band alignment, significantly suppresses carrier recombination rates and facilitate efficient separation and transfer of carriers, thereby leading to the enhanced PEC performance in heterojunction sample. This study provides a promising approach for the practical application development of high-performance PEC systems.

Data availability

All relevant data are available in the manuscript and the ESI.†

Conflicts of interest

There are no conflicts to declare in this paper.

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References

- 1 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- 2 H. Tong, S. X. Ouyang, Y. Bi, N. Umezawa, M. Oshikiri and J. H. Ye, *Adv. Mater.*, 2012, **24**, 229–251.
- 3 J. Q. Lv, J. F. Xie, A. G. A. Mohamed, X. Zhang and Y. B. Wang, *Chem. Soc. Rev.*, 2022, **51**, 1511–1528.
- 4 Y. X. Tan, X. Zhang, J. Lin and Y. B. Wang, *Energy Environ. Sci.*, 2023, **16**, 2432–2447.
- 5 Z. R. Wang and H. Einaga, *ChemCatChem*, 2023, **15**, e202300723.
- 6 M. A. Gaikwad, U. P. Suryawanshi, U. V. Ghorpade, J. S. Jang, M. P. Suryawanshi and J. H. Kim, *Small*, 2022, **18**, 2105084.
- 7 F. Li, J. Jian, Y. X. Xu, S. Y. Wang, H. Y. Wang and H. Q. Wang, *Eng. Rep.*, 2021, **3**, e12387.
- 8 B. Zutter, Z. J. Chen, L. Barrera, W. Gaieck, A. S. Lapp, K. Watanabe, A. Kudo, D. V. Esposito, R. B. Chandran, S. Ardo and A. A. Talin, *ACS Nano*, 2023, **17**, 9405–9414.
- 9 C. Zhen, X. T. Chen, R. T. Chen, F. T. Fan, X. X. Xu, Y. Y. Kang, J. D. Guo, L. Z. Wang, G. Q. Lu, K. Domen, H. M. Cheng and G. Liu, *Nat. Commun.*, 2024, **15**, 1672.
- 10 K. Iwashina and A. Kudo, *J. Am. Chem. Soc.*, 2011, **133**, 13272–13275.
- 11 S. Kawasaki, K. Nakatsuji, J. Yoshinobu, F. Komori, R. Takahashi, M. Lippmaa, K. Mase and A. Kudo, *Appl. Phys. Lett.*, 2012, **101**, 033910.
- 12 M. Antuch, P. Millet, A. Iwase and A. Kudo, *Electrochim. Acta*, 2019, **297**, 696–704.
- 13 R. Konta, T. Ishii, H. Kato and A. Kudo, *J. Phys. Chem. B*, 2004, **108**, 8992–8995.
- 14 S. Kawasaki, K. Akagi, K. Nakatsuji, S. Yamamoto, I. Matsuda, Y. Harada, J. Yoshinobu, F. Komori, R. Takahashi, M. Lippmaa, C. Sakai, H. Niwa, M. Oshima, K. Iwashina and A. Kudo, *J. Phys. Chem. C*, 2012, **116**, 24445–24448.
- 15 D. H. K. Murthy, H. Matsuzaki, Q. Wang, Y. Suzuki, K. Seki, T. Hisatomi, T. Yamada, A. Kudo, K. Domen and A. Furube, *Sustainable Energy Fuels*, 2019, **3**, 208–218.
- 16 M. Guo and G. J. Ma, *J. Catal.*, 2020, **391**, 241–246.
- 17 S. Okunaka, H. Tokudome and R. Abe, *Chem. Lett.*, 2016, **45**, 57–59.
- 18 R. P. Sugavaneshwar, K. Chen, G. Lakshminarayana, S. Ishii, T. D. Dao, N. Umezawa and T. Nagao, *APL Mater.*, 2015, **3**, 116103.
- 19 S. Zhang, E. H. Jiang, J. Wu, Z. H. Liu, Y. Yan, P. W. Huo and Y. S. Yan, *Catalysts*, 2023, **13**, 851.
- 20 D. Yazaki, T. Kawawaki, T. Tanaka, D. Hirayama, Y. Shingyouchia and Y. Negishi, *Energy Adv.*, 2023, **2**, 1148–1154.
- 21 B. Y. Zhang, K. W. Liu, Y. Xiang, J. M. Wang, W. R. Lin, M. Guo and G. J. Ma, *ACS Catal.*, 2022, **12**, 2415–2425.
- 22 W. J. Ong, L. L. Tan, Y. H. Ng, S. T. Yong and S. P. Chai, *Chem. Rev.*, 2016, **116**, 7159–7329.
- 23 A. Akhundi, A. Z. Moshfegh, A. H. Yangjeh and M. Sillanpää, *ACS ES&T Eng.*, 2022, **4**, 564–585.
- 24 L. X. Wang, Y. L. Dong, J. Y. Zhang, F. F. Tao and J. J. Xu, *J. Solid State Chem.*, 2022, **308**, 122878.
- 25 W. He, L. Liu, T. T. Ma, H. M. Han, J. J. Zhu, Y. P. Liu, Z. Fang, Z. Yang and K. Guo, *Appl. Catal., B*, 2022, **306**, 121107.
- 26 E. M. Hu, Q. Liu, Z. S. Qian, Q. Zhong, J. H. He, S. C. Xu, T. M. Lu, J. Li, T. Chen and W. K. Zhu, *Small*, 2024, **43**, 2403105.
- 27 Q. C. Feng, Z. Liu, R. D. Su, Y. Chen, Y. Wang, D. F. Ma and Q. Li, *Appl. Catal., B*, 2025, **362**, 124714.
- 28 B. Wang, P. Li, H. J. Hao, H. J. He, H. R. Cai, F. F. Shang, B. An, X. Q. Li and S. C. Yan, *Nanomaterials*, 2022, **12**, 4428.
- 29 A. Kumar, V. N. Rao, A. Kumar, A. Mushtaq, L. Sharma, A. Halder, S. K. Pal, M. V. Shankar and V. Krishnan, *ACS Appl. Energy Mater.*, 2020, **3**, 12134–12147.
- 30 M. Antuch, P. Millet, A. Iwase, A. Kudo, S. A. Grigoriev and Y. Z. Voloshin, *Electrochim. Acta*, 2017, **258**, 255–265.
- 31 L. L. Wang, G. G. Tang, S. Liu, H. L. Dong, Q. Q. Liu, J. F. Sun and H. Tang, *Chem. Eng. J.*, 2022, **428**, 131338.
- 32 J. Chen, M. G. Wang, J. Han and R. Guo, *J. Colloid Interface Sci.*, 2020, **562**, 313–321.
- 33 M. Khandelwal, C. V. Tran and J. B. In, *Appl. Surf. Sci.*, 2022, **576**, 151714.
- 34 M. Khandelwal, C. V. Tran, J. Lee and J. B. In, *Chem. Eng. J.*, 2022, **428**, 131119.
- 35 X. X. Zhao, Y. H. Sang, H. Liu and X. W. Yu, *Adv. Sustainable Syst.*, 2024, 2400756.
- 36 H. S. Lu, S. X. Song, Q. S. Jia, G. B. Liu and L. H. Jiang, *Acta Phys.-Chim. Sin.*, 2024, **40**, 2304035.
- 37 J. X. Low, J. G. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, Heterojunction Photocatalysts, *Adv. Mater.*, 2017, **29**, 1601694.
- 38 S. X. Zhong, Y. M. Xi, Q. Chen, J. R. Chen and S. Bai, *Nanoscale*, 2020, **12**, 5764–5791.
- 39 D. W. Cao, M. Li, J. F. Zhu, Y. F. He, T. Chen, Y. Liu, M. M. Chen and Y. Yang, *J. Semicond.*, 2021, **42**, 112701.
- 40 S. Wang, F. G. Zheng, Y. Y. Weng, G. L. Yuan, L. Fang and L. You, *Adv. Mater. Interfaces*, 2020, **10**, 2000185.
- 41 A. R. C. Bredar, A. L. Chown, A. R. Burton and B. H. Farnum, *ACS Appl. Energy Mater.*, 2020, **3**, 66–98.
- 42 A. Ch. Lazanas and M. I. Prodromidis, *ACS Meas. Sci. Au*, 2023, **3**, 162–193.
- 43 X. L. Liao, T. T. Li, H. T. Ren, Z. Y. Mao, X. F. Zhang, J. H. Lin and C. W. Lou, *Ceram. Int.*, 2021, **47**, 10786–10795.
- 44 Y. Z. Liu, S. S. Ding, J. Xu, H. Y. Zhang, S. G. Yang, X. G. Duan, H. Q. Sun and S. B. Wang, *Chin. J. Catal.*, 2017, **38**, 1052–1062.
- 45 Y. Xu and M. A. A. Schoonen, *Am. Mineral.*, 2000, **85**, 543–556.
- 46 S. Vignesh and H. Kim, *J. Alloys Compd.*, 2023, **942**, 169077.
- 47 X. Zhao, M. J. Liu, Y. C. Wang, Y. Xiong, P. Y. Yang, J. Q. Qin, X. Xiong and Y. P. Lei, *ACS Nano*, 2022, **16**, 19959–19979.

