



Cite this: *RSC Adv.*, 2025, **15**, 7181

Study about the effect of cellulose nanocrystals on a polyacrylate miniemulsion

Zeping Wang,^a Lionel O'Young,^a Sajid Mahmood,^{ab} George Zheng Chen,^b Yitao Zheng^d and Binjie Hu^{*d}

Cellulose nanocrystals (CNC) are widely used due to their biodegradability, high strength, large surface area, and functional versatility. This study investigates the interaction between CNC and acrylate emulsions, which mainly focuses on their impact on emulsion characteristics, polymerization behaviour, and storage stability. CNC was incorporated into an acrylate miniemulsion system at varying concentrations, followed by the systematic study of its effects on particle size, interfacial tension, zeta potential, yield, and viscosity. The morphology of CNC-acrylate systems was analysed using infrared spectroscopy and scanning electron microscopy (SEM). The results demonstrated that CNC effectively co-stabilized acrylate miniemulsions and enhanced their stability before polymerization. Although CNC did not directly participate in polymerization or affect yield or reaction rates, it slowed the diffusion of free radicals. However, CNC concentrations higher than 1 wt% negatively impacted post-polymerization storage stability and caused aggregation of droplets. These findings reveal the dual role of CNC as both a stabilizing and aggregating agent, offering new insights into its potential for the design of advanced polymer systems.

Received 13th September 2024
 Accepted 27th February 2025

DOI: 10.1039/d4ra06614f

rsc.li/rsc-advances

1. Introduction

In the coating industry, there is a potential to transform through sustainability, improved performance, and multi-functionality. Waterborne coatings are increasingly favoured due to their reduced environmental impact in lower volatile organic compound (VOC) emissions an alignment with global sustainability goals and regulatory requirements.¹ These coatings utilize water as the primary solvent, thus offering an eco-friendly solution. However, challenges like suboptimal mechanical strength, stability in waterborne systems have necessitated the development of innovative additives to improve performance.²

To solve those challenges, seeking more novel potential additive materials could be a solution. Cellulose as the most abundant biopolymer on earth, has drawn more attention recently on its various additive function due to its ease of

altering, tailoring and manufacture.^{3,4} Cellulose nanocrystal (CNC), a rod-shaped nanomaterial extracted from cellulose microfibrils, has gained significant attention for their unique properties like high crystallinity and strong mechanical properties, and diverse applications.⁵ Compared with cellulose nanofiber (CNF), CNC is shorter in length and has a higher ratio of crystalline structure, exhibiting high specific strength, modulus, large surface area, gas impermeability, and unique liquid crystalline properties.^{6,7} In addition, CNC also shares advantages of cellulose, which are biodegradable, biocompatible, renewable and environmentally friendly.⁸ Attracted by these characteristics, CNC is widely applied in nanocomposites films and membranes for food packaging,⁹ conductive materials,¹⁰ water treatment,¹¹ biomedical use,¹² and waterborne coatings,^{13–16} especially. Moreover, the enhancement of CNC on film strength, water resistance, and environmental sustainability was also proven inside a waterborne polyacrylate system in architectural, automotive, and anticorrosive coatings.^{17–19}

Despite these developments on CNC, previous research mainly focused on the applications of CNC, not much systematic research has been carried out. Especially the mechanisms and behaviour of CNC within coatings is yet understood. To address this gap, this study systematically investigates the effects of CNCs within a waterborne polyacrylate coating, providing deeper insights into their role and interactions between CNC and polymers within the coating.

^aGreen Chemicals & Energy Process Development Laboratory, China Beacons Institute, University of Nottingham Ningbo China, 199 Taikang East Road, Ningbo, 315000, China

^bLow Dimensional Materials Research Center, Khazar University, Baku, AZ1096, Azerbaijan

^cDepartment of Chemical and Environmental Engineering, Advanced Materials Research Group, Faculty of Engineering, The University of Nottingham, Nottingham, NG7 2RD, UK

^dDepartment of Chemical & Environmental Engineering, University of Nottingham Ningbo China, 199 Taikang East Road, Ningbo, 315000, China. E-mail: Binjie.HU@nottingham.edu.cn



2. Experimental methods and materials

2.1. Materials

Methyl methacrylate (MMA, CP), sodium dodecyl sulfonate (SDSO, CP), L-ascorbic acid (AAc, AR) and hydrogen peroxide (H_2O_2 , AR) were supplied by Sinopharm Chemical Reagent Co., Ltd. Hexadecane (HD, 98%) and *n*-butyl acrylate (BA, 99%) were offered by Aladdin Industrial Inc. Cellulose nanocrystal (CNC) was purchased from ScienceK Co., Ltd. Ultra-pure water (18.2 MΩ cm) was obtained from a ultrapure water system (Milli-Q® IQ 7000).

2.2. Preparation of waterborne acrylate coating

2.2.1. Preparation of oil phase. The oil phase was mixed with HD (5.0 wt%), MMA (47.5 wt%) and BA (47.5 wt%) under magnetic mixing at 200 rpm, where HD was used for the prevention of Ostwald ripening.

2.2.2. Preparation of water phase. The water phase was formed with water, SDSO (30 mM) and CNC (0, 0.1, 0.5, 1, 1.5 and 2 wt%) under magnetic stirring for 30 minutes.

2.2.3. Homogenization of oil phase and water phase. The oil phase and water phase were homogenized in a sonicator (Xinzhi Scientz II) for 6 minutes (1 second on and 1 second off) with oil-in-water (O/W) volume ratio of 3 : 7 at 40 °C and 285 W.

2.3. Miniemulsion polymerization

After homogenization of oil phase and water phase, the mixture was loaded into a 250 mL four-necked flask equipped with a stirrer, a reflux condenser, a thermometer, and a nitrogen inlet. The temperature was controlled at 40 °C with a water bath. After stirring at 200 rpm and nitrogen bubbling for 1.5 h to remove oxygen, H_2O_2 /AAc solution (0.1 mol% of the oil phase) was introduced to initiate the polymerization.

2.4. Characterization techniques

2.4.1. Droplet size. The droplet sizes and its size distribution were measured with a laser diffraction method (Mastersizer 3000, Malvern Inc).

2.4.2. Interfacial tension. The interfacial tension at the oil/water interface was measured using the Du Noüy sheet method (BZY-2 tensiometer, Hengping Instruments).

2.4.3. Zeta potential. Zeta potentials of the acrylate miniemulsion were measured with a trace laser doppler electrophoresis method (ZetasizerNano ZS, Malvern Inc.) at room temperature. The sample was diluted one hundred times with water and pH of each sample was controlled at 5 to prevent the pH interference. For each sample, the measurement was repeated for three times.

2.4.4. SEM. The morphology of polyacrylate samples with or without CNC was studied by the scanning electron microscopy (SEM) (RIGMA/VP, Carl Zeiss Microscopy Ltd) under a 3 kV accelerating voltage. The polyacrylate emulsions were diluted one thousand times, dropped on a silicon slice, dried in air and placed at a platform for observation.

2.4.5. Yield. The yield of PMMA-*co*-PBA was calculated gravimetrically due to the volatile nature of monomers and hydrophobes. Samples were simultaneously taken at regular intervals and stored in an ice bath to terminate reaction. After that, samples were placed in a fume cupboard to evaporate the monomer and hydrophobe residue. Then samples were baked in a vacuum oven at 120 °C for 4 h to remove moisture. The yield of polymers could be obtained from the mass ratio of polymer and monomer.

2.4.6. Viscosity. Viscosities of miniemulsion after polymerization were measured using the rotational rheometer with a plate-and-plate system (Kinexus Pro+, Malvern Inc), where the diameter of the upper plate is 40 mm and the gap between the upper and lower plates is 1.0 mm. Moreover, the measurement was conducted at a shear rate ranging from 0.1 s⁻¹ to 1000 s⁻¹ at room temperature (25 °C).

2.4.7. FTIR. FTIR spectra of polyacrylate samples with or without CNC was collected from the Bruker vertex 70 instrument.

3. Results and discussion

3.1. Effect of the CNC on the o/w interface of acrylate miniemulsion

The polyacrylate emulsion was composed of an oil phase, a water phase, and surfactants, which was proven to be an effective coating formula in our previous studies.^{20,21} Initially, to study the effect of CNC on the o/w interface, the interfacial tension between the oil and water phase at different concentrations of CNC without SDSO was measured and presented in Fig. 1. The interfacial tension remarkably decreased from 14.2 mN m⁻¹ to 10.2 mN m⁻¹ after introducing 0.1 wt% CNC. The value further decreased to 9.2 mN m⁻¹ as the CNC content increased to 1 wt%, then remained relatively constant with further increases in CNC content up to 2 wt%. This validated that the CNC could effectively reduce the interfacial tension between the oil and water phase. However, the mixture of oil

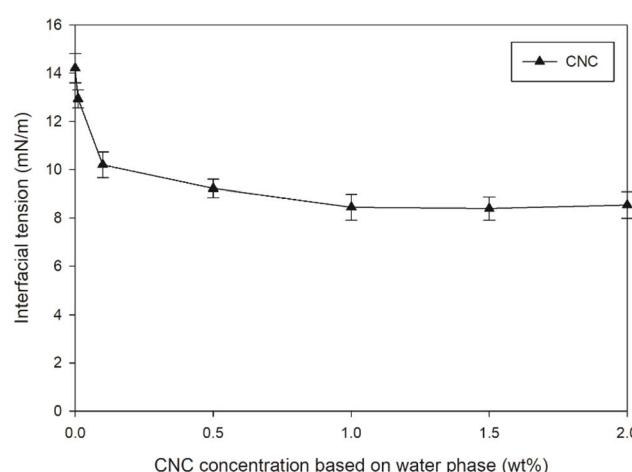


Fig. 1 Variation in interfacial tension at different CNC contents in water phase (without SDSO).



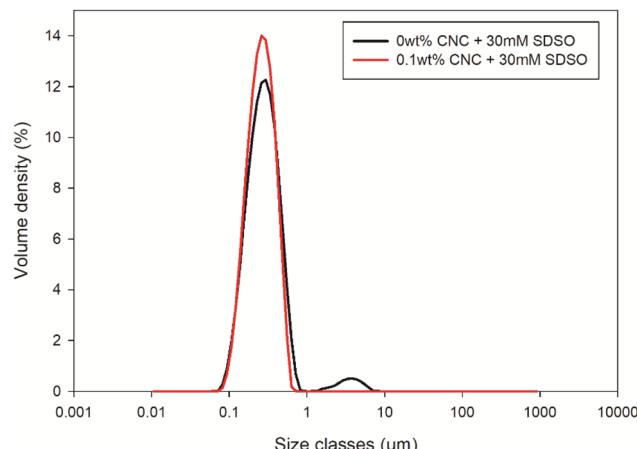


Fig. 2 Variation in initial droplet size distribution with or without 0.1wt% CNC in water phase (30 mM SDSO).

and water phases after ultrasonication could not achieve a homogeneous emulsion without SDSO. In other words, the CNC as received could not fully stabilize the o/w interface. Therefore, SDSO was introduced as the main stabilizer into this formulation. 30 mM was the cost-effective concentration for SDSO in an emulsion.²²

After adding 30 mM SDSO, the emulsion was supposed to be homogenous without CNC. However, there was still a small portion of the emulsion remained unstable. As shown in Fig. 2, two peaks were observed in the size distribution plot of emulsion without CNC, where the main peak distributed in the range of 0.1–1 μm, and the other small peak located between 1–10 μm. The main peak represented the stable emulsion, while the smaller peak could be ascribed to the flocculation or partial coalescence of a small quantity of oil droplets.^{23,24} In comparison, the emulsion with 0.1 wt% CNC presented only a single, tight peak in the size distribution plot, which indicated that the introduced CNC could effectively prevent flocculation or partial coalescence of oil droplet. In other words, CNC could act as a co-stabilizer in this coating formula. Therefore, the Sauter mean

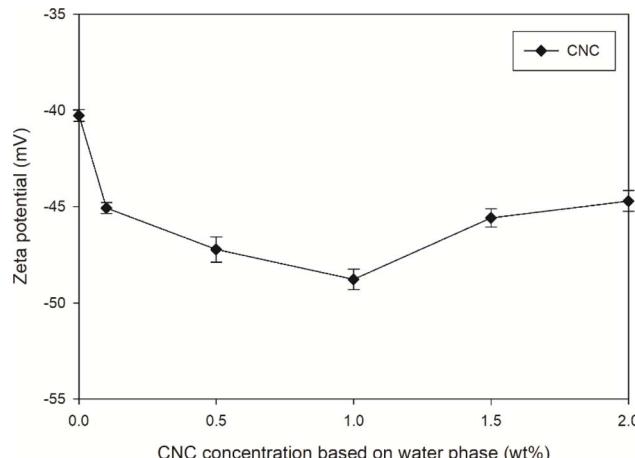


Fig. 4 Effect of CNC content on the zeta potential (30 mM SDSO).

diameter ($D_{3,2}$) of oil droplet decreased from 248 nm to 235 nm after introducing 0.1 wt% CNC.

To prove the co-stabilization effect of CNC, the interfacial tension and droplet size at different CNC contents with 30 mM SDSO was measured and presented in Fig. 3. The interfacial tension reduced from 3.7 mN m⁻¹ to 3.4 mN m⁻¹ after the CNC content reaching 0.1 wt%. With increasing the CNC content to 1 wt%, the interfacial tension further reduced to 3.3 mN m⁻¹, despite that the reduction was limited. Similar reductions in the interfacial tension were recorded by Hu, *et al.*²⁵ which was attributed to the intermediate wettability of CNC. The interfacial tension and particle size were positively correlated.²⁶ This was proven by the change of initial droplet size in Fig. 3. More particles could nucleate at lower interfacial tension.²⁷ For a fixed volume of oil phase, more particles indicated the smaller size in single droplet.

CNC was reported as a Pickering agent to stabilize the oil-water interface in emulsions due to intermediate wettability and nanometric size.^{28–32} Pickering emulsions were also known as particle-stabilized emulsions. Like other Pickering agents, CNC could prevent coalescence and sedimentation by forming a tightly packed layer.³³ Although there was still a debate on whether particles could also decrease the interfacial tension like surfactants.³⁴ Some literature insisted claiming that they found the reduction of interfacial tension with particles as stabilizers.^{35–39}

To further validate the co-stabilization effect, zeta potential of each sample was measured and presented in Fig. 4. Empirically, the emulsion was considered as a stable system if the potential values higher than +30 mV or lower than -30 mV, which indicated that particles repelled each other and form a stable dispersion.⁴⁰ Moreover, a higher absolute value of zeta potential referred to a higher stable state of colloidal systems.⁴¹ In this way, all the samples with or without CNC were stable. After introducing 0.1 wt% CNC, the absolute value of zeta potential increased from 40.2 to 45.1 mV, indicating the stability enhancement. This is another evidence of CNC to be the co-stabilizer.

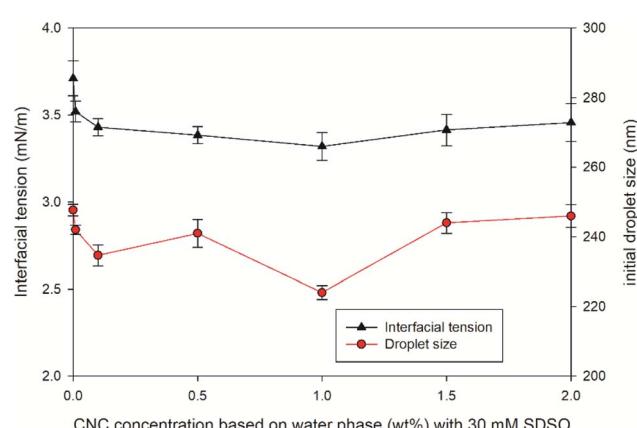


Fig. 3 Effect of CNC content on the initial droplet size and interfacial tension between oil and water phase (30 mM SDSO).



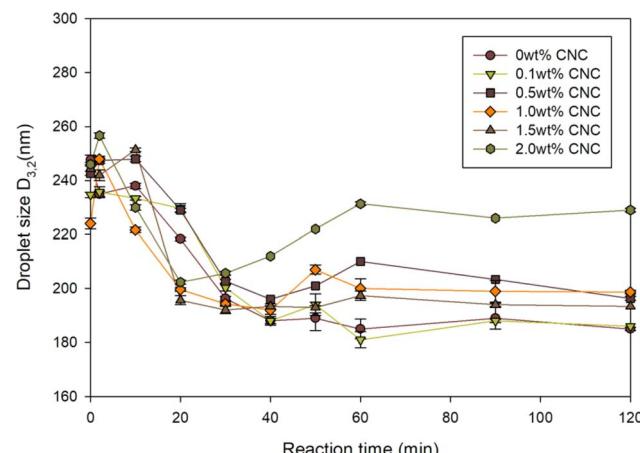


Fig. 5 Droplet size of each sample during the reaction as a function of CNC content.

3.2. Effect of the CNC on the acrylate miniemulsion polymerization

3.2.1. Effect of CNC on miniemulsion polymerization. After preparing the miniemulsion, the effect of CNC on droplet size and reaction during miniemulsion polymerization was investigated. As shown in Fig. 5, $D_{3,2}$ of various samples with different CNC contents varied over time. After introducing initiators, $D_{3,2}$ of all samples exhibited a slight rise due to temporary instability. Unlike emulsion polymerization, droplet nucleation was the predominant nucleation mode.⁴² In other words, droplets nucleated and formed polymer particles. Except 2 wt% CNC sample, the reduction of $D_{3,2}$ during the first hour could be attributed to the formation of polymer particles. Moreover, the relatively constant $D_{3,2}$ in the second hour indicated the termination of miniemulsion polymerization. However, among these samples, the size change of 2 wt% CNC sample differed. $D_{3,2}$ of 2 wt% CNC sample presented an increasing trend at first 20 min, which could be ascribed to the depletion flocculation of emulsion.⁴³ Depletion flocculation occurred in oil-in-water emulsions when the concentration of non-adsorbed

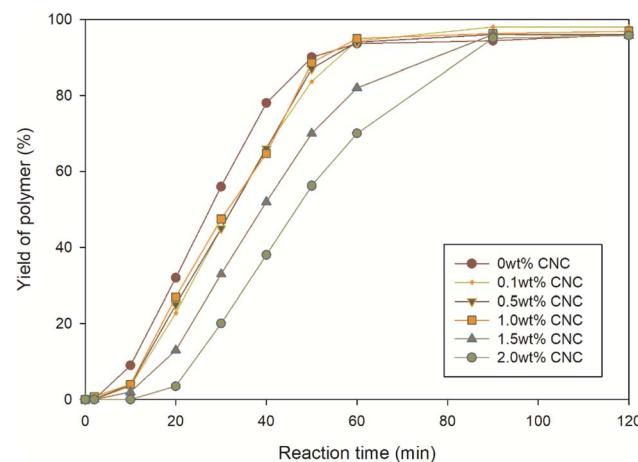


Fig. 7 Variation in yield as a function of CNC content.

polysaccharide exceeded a certain level. In this study, CNC was a type of polysaccharide. When the CNC content was higher than 2 wt%, instability occurred and caused the aggregation or coagulation of droplets. To prove this, the viscosity of various samples at different shear rates was measured and displayed in Fig. 6. All the samples presented the shear thinning behaviour, particularly at 2 wt%. The viscosity at 0.1 s^{-1} was 100 times higher than the one of the sample without CNC, which indicated aggregation or coagulation of droplets breaking with increasing shear rate.⁴⁴

In addition to the droplet size, the yield of miniemulsion polymerization at different CNC contents was calculated and displayed in Fig. 7. The yield of each sample was similar, approximately 96% at the end of the experiment. Most of samples achieved the highest yield within first hour, except for the 1.5 wt% and 2 wt% CNC samples. The difference in termination time might be affected by the rate of miniemulsion polymerization R_p and diffusion coefficient D of initiator from water phase to monomer droplet. Bechthold and Landfester⁴⁵ proposed an equation to calculate miniemulsion polymerization rate R_p (eqn (1)).

$$R_p = \frac{k_p[M]\bar{n}N}{N_A} \quad (1)$$

where k_p was the reaction rate constant, $[M]$ was the monomer concentration, \bar{n} was the average effective radical number per droplet, N was the number of droplets per liter of water phase, and N_A was the Avogadro's number.

In this study, the reaction temperature, monomer and initiator amount concentration were fixed, so the value of k_p , $[M]$ and \bar{n} remained constant. As for the number of droplets per liter of water phase N , it could be calculated with eqn (2).

$$N = \frac{V_{\text{oil}}}{\frac{4}{3}\pi r^3 V_{\text{water}}} \quad (2)$$

where r was droplet radius, V_{oil} and V_{water} represented the volumes of the oil phase and water phase, respectively. The volume ratio of the oil and water phase was identical in this

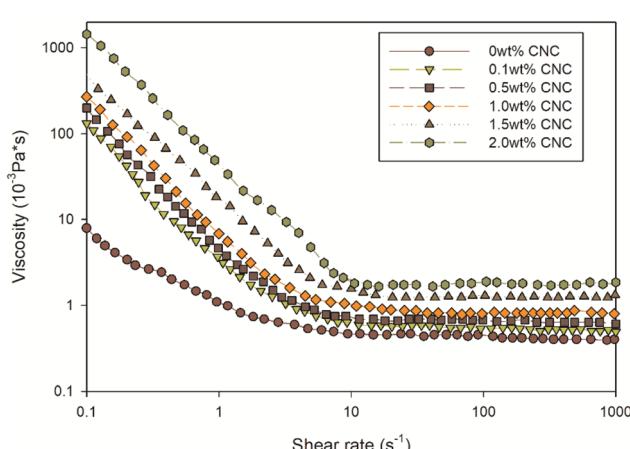


Fig. 6 Variation in viscosity versus shear rate as a function of CNC content.



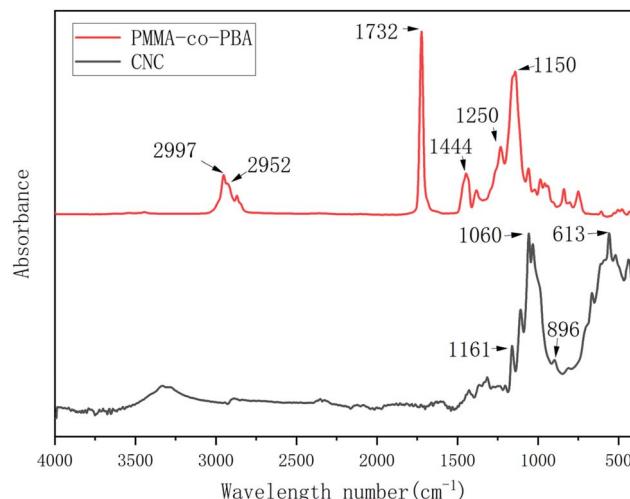


Fig. 8 FTIR spectra for PMMA-co-PBA and CNC.

study. The only difference was the radius of droplet, which ranged from 225 to 250 nm, still within the same size range. Therefore, the curves of linear increasing phase were parallel to each other and R_p of each sample was similar.

However, the reaction initiation point was different, which could be influenced by the diffusion of free radicals from initiator to monomer. The Stokes-Einstein equation in rotational diffusion⁴⁶ was given as listed,

$$D = \frac{k_B T}{8\pi\eta r^3} \quad (3)$$

where D was diffusion coefficient, k_B was the Boltzmann's constant, T was absolute temperature, η was dynamic viscosity and r was radius of the spherical particle. In this equation, diffusion coefficient D was inversely related to viscosity η and the radius r of the moving particles. As shown in Fig. 6, the viscosity increased with the increment of CNC loading. Therefore, it would take longer time for free radicals to transport into samples with higher CNC content. And after that, once the free radicals reached the reaction site and miniemulsion polymerization initiated, all the samples presented the similar reaction rate.

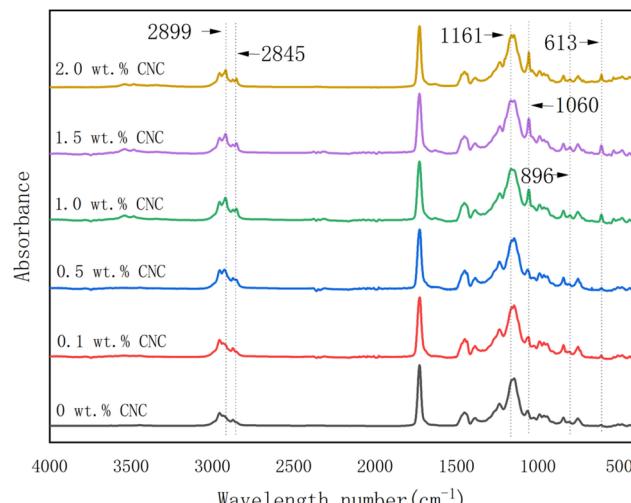


Fig. 9 FTIR spectra for polyacrylate samples containing different CNC contents.

3.2.2. Effect of CNC on interactions with PMMA-*co*-PBA inside the polyacrylate coating. In addition to the effect of CNC on miniemulsion polymerization, interactions of CNC between PMMA-*co*-PBA were another area of investigation. Motivated by this, CNC as received and polyacrylate samples with different CNC contents were analysed using FTIR. Firstly, the FTIR of CNC and PMMA-*co*-PBA were measured and presented in Fig. 8. For PMMA, peaks at 2997 cm^{-1} , 2952 cm^{-1} and 1444 cm^{-1} were assigned to the C-H stretching vibrations of $-\text{CH}_3$ and $-\text{CH}_2-$ groups, and the C-H bending vibration of $-\text{CH}_3$ and $-\text{CH}_2-$ groups, respectively.^{47,48} Peaks at 1732 cm^{-1} , 1250 cm^{-1} and 1150 cm^{-1} represented the stretching vibrations of $\text{C}=\text{O}$ and $\text{C}-\text{O}$ in ester group, respectively, validating the existence of PMMA-*co*-PBA.^{49,50} For CNC, the peak at 613 cm^{-1} was caused by the stretching vibration of aromatic $-\text{CH}$.⁵¹ The peak at 896 cm^{-1} could be the characteristics of the $\text{C}-\text{O}-\text{C}$ symmetric stretching vibration of β -glycosidic linkages.⁵² Additionally, peaks at 1161 cm^{-1} and 1060 cm^{-1} corresponded to the stretching vibration of $\text{C}-\text{O}-\text{C}$ in the pyranose ring and the bending vibration of $\text{C}-\text{O}-\text{C}$ in β -glycosidic linkages, respectively.⁵³⁻⁵⁶ For the ease of comparison, these characteristic peaks were summarized and listed in Table 1.

Table 1 Peak assignment for CNC-incorporated PMMA-*co*-PBA

Peak location (cm^{-1})	Origin of peak	Reference
2997	C-H stretching vibration of $-\text{CH}_3$ groups	47
2952	C-H stretching vibration of $-\text{CH}_2-$ groups	47
2899	C-H asymmetric stretching vibration in glucose units	56 and 57
2845	C-H symmetric stretching vibration in glucose units	57
1732	$\text{C}=\text{O}$ stretching vibration	49 and 50
1444	C-H bending vibration of $-\text{CH}_3$ and $-\text{CH}_2-$ groups	48
1250, 1150	$\text{C}-\text{O}$ stretching vibrations of ester groups	49 and 50
1161	$\text{C}(\text{O})-\text{O}$ stretching vibration in pyranose ring	53-55
1060	$\text{C}-\text{O}-\text{C}$ bending vibration of β -glycosidic linkages	56
896	$\text{C}-\text{O}-\text{C}$ symmetric stretching vibration of β -glycosidic linkages	52
613	Aromatic $-\text{CH}$ stretching vibration	51



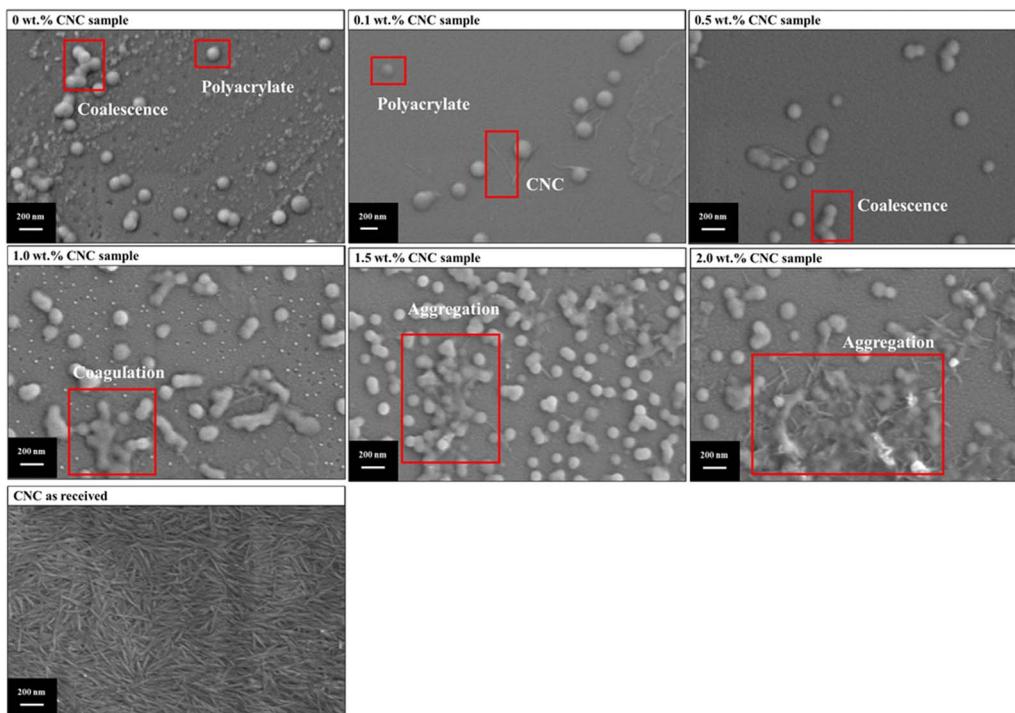


Fig. 10 SEM for polyacrylate samples containing different CNC contents and CNC as received.

As shown in Fig. 9, FTIR spectra of polyacrylate samples with different CNC contents were presented. In the spectra, the peaks at 1161, 1060, 896, and 613 cm^{-1} became more apparent when CNCs content exceeded 0.5 wt%, which indicated the successful addition of CNC. Moreover, peaks at 2899 cm^{-1} and 2845 cm^{-1} were also observed in the samples with higher CNC contents (>0.5 wt%). These two peaks were categorized to C–H asymmetric and symmetric stretching vibration in glucose units, respectively.^{56,57} Meanwhile, CNC contained glucose units. In other words, this was another evidence that CNC was successfully incorporated into the polyacrylate samples. However, no other new peaks were observed, which indicated that no new substance formed. Therefore, CNC was estimated to exist between PMMA-*co*-PBA without attending the reaction.

To further evaluate the interaction between CNC and PMMA-*co*-PBA, the morphology of polyacrylate samples with or without CNC and CNC as received was investigated with SEM, which was shown in Fig. 10. For the polyacrylate sample without CNC, both the single droplet and coalesced droplets could be observed. After adding 0.1 wt% CNC, the polyacrylate sample was dispersed, and CNC located near the droplet. Similar phenomenon was observed in the 0.5 wt% sample, while the coalescence of droplets was more severe. With further increasing the CNC content to 1 wt%, the coagulation happened. Then the aggregation was found in 1.5 wt% and 2 wt% samples, where the aggregation of 2 wt% is relatively harder.

The interaction between CNC and PMMA-*co*-PBA was further evaluated using SEM to analyse the morphology of polyacrylate samples with varying CNC concentration (Fig. 10). Fig. 10g

confirmed the rod-like shape of CNC. In the absence of CNC. In the absence of CNC (Fig. 10a), the polyacrylate sample exhibited the coexist of individual droplets and droplets coalescence, which suggests the instability in such system. Adding 0.1 wt% CNC (Fig. 10b) resulted in noticeable uniform dispersion within the polymer matrix, with CNC particles predominantly localized near the droplets. This suggested that presence of CNC at lower concentrations could prevent droplets coalescence even. This further suggests that CNC could act as a stabilizing agent in emulsion formation. However, further increase CNC concentration to 0.5 wt%, the droplet coalescence became more pronounced. As the CNC content increased to 1 wt%, coagulation was observed. At higher CNC concentrations (1.5 wt% and 2 wt%), significant aggregation occurred, with the most severe aggregation seen in the 2 wt% sample. This phenomenon the function of CNC as co stabiliser is limited at lower concentration. At higher concentration, CNC particle–particle interactions become dominant to induce droplets coalescence hence jeopardise stability of emulsion.⁵⁸

These results highlighted the dual role of CNC as both a stabilizing and aggregating agent, depending on its concentration. As shown in Fig. 11, the interaction between CNC and PMMA-*co*-PBA was illustrated. Without CNC, the acrylate emulsion can be initiated by free radicals to polyacrylate emulsion with reducing droplet size, proven by the droplet size change in Fig. 5. The addition of suitable amount of CNC can help the dispersion of acrylate and polyacrylate emulsion. After introducing excessive amount of CNC, the dispersion of acrylate droplets in emulsion might still be enhancing, proven by the droplet size, interfacial tension (Fig. 3), and zeta potential



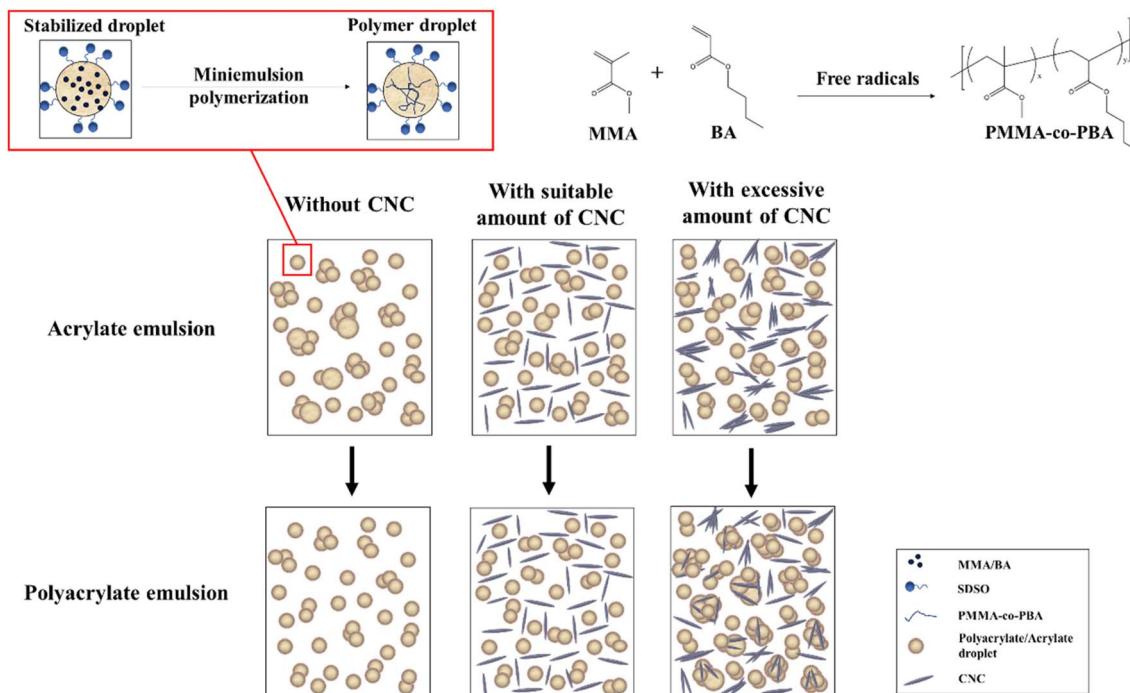


Fig. 11 Conceptual diagram of the interaction between CNC and PMMA-co-PBA.

(Fig. 4). However, the stability of polyacrylate emulsion would not be as stable as the one without CNC, coagulation and aggregation happened during the polymerization and influenced the storage stability of emulsion.

3.3. Effect of the CNC on storage stability of polyacrylate miniemulsion

As discussed in Section 3.2.2, the storage stability of polyacrylate miniemulsion might be influenced by the CNC contents. According to Anton and Vandamme,⁵⁹ changes in droplet size over time could indicate the storage stability of the emulsion. To observe the effect of CNC on the storage stability of emulsion, the droplet size of acrylate miniemulsions was measured every 7 days. Although the miniemulsion appeared visually stable, droplet sizes shown in Fig. 12 increased after 4 weeks of storage at room temperature. For the 0 wt% CNC sample, the miniemulsion was relatively stable, with initial and final droplet sizes of 192 nm and 198 nm, respectively. Similarly, the samples with CNC content lower than 0.5 wt% were also stable. This might be caused by the interactions between anionic surfactants and CNC.⁶⁰ Without CNC, the emulsion was only stabilized by SDSO with hydrophobic site in the oil phase and hydrophilic site in the water phase. After introducing CNC, it might cooperate with SDSO, where SDSO stabilized the O/W interface, and CNC formed a stable network under hydrogen bonding and electrostatic repulsion.^{61,62} However, when the content exceeded 1 wt%, CNC would apply the adverse effect on the storage stability of the miniemulsion. Especially for 2 wt% CNC sample, the droplet size increased significantly from 202 nm to 230 nm, which indicated the coagulation and

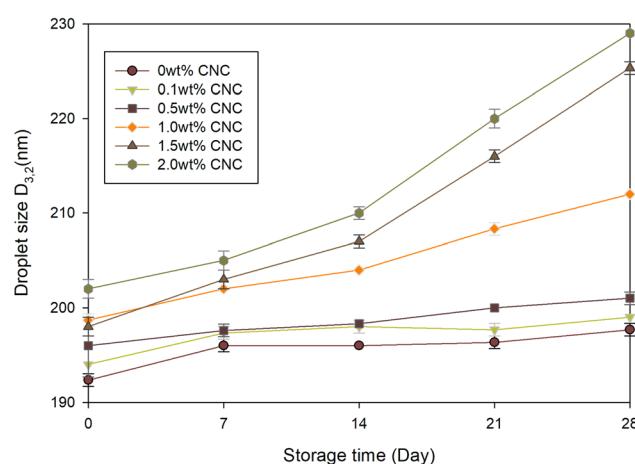


Fig. 12 Variation in droplet size as a function of storage time.

aggregation of droplets.⁶³ This was proven by the SEM images (Fig. 10) as a sign of deterioration in storage stability.

4. Conclusion

This study offers a comprehensive analysis of the interactions between CNC and PMMA-co-PBA in an emulsion system, both before and after polymerization. It provides valuable insights into CNC's functional roles. The results demonstrate that CNC effectively co-stabilizes acrylate miniemulsions before polymerization. While CNC did not directly participate in polymerization or affect yield or reaction rates, it slowed the diffusion of free radicals during the process. For certain



concentrations, such as 1 wt%, CNC enhanced emulsion stability before polymerization but negatively impacted storage stability afterward. These findings highlight the dual role of CNC as both a stabilizing and aggregating agent, emphasizing its potential for tailoring polymer properties for industrial applications. Future research should focus on modifying CNC to improve its stabilizing effects on both the initial emulsion and the final polymerized system, while further elucidating the underlying mechanisms of its interactions with polymer matrices.

Data availability

The data are available from the corresponding author on reasonable request.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The research is financially supported by Zhejiang Yuxi Corrosion Control Corporation, grant number (01.03.07.01.2015.06.002).

References

- 1 A. Arjmandi, H. Bi, S. U. Nielsen and K. Dam-Johansen, From Wet to Protective: Film Formation in Waterborne Coatings, *ACS Appl. Mater. Interfaces*, 2024, **16**, 58006–58028.
- 2 H. Yao, L. Li, W. Li, D. Qi, W. Fu and N. Wang, Application of nanomaterials in waterborne coatings: A review, *Resour. Chem. Mater.*, 2022, **1**(2), 184–200.
- 3 T. Li, *et al.*, Developing fibrillated cellulose as a sustainable technological material, *Nature*, 2021, **590**(7844), 47–56.
- 4 G. Jiang, *et al.*, A scalable bacterial cellulose ionogel for multisensory electronic skin, *Research*, 2022, DOI: [10.34133/2022/9814767](https://doi.org/10.34133/2022/9814767).
- 5 F. Garavand, M. Nooshkam, D. Khodaei, S. Yousefi, I. Cacciotti and M. Ghasemlou, Recent advances in qualitative and quantitative characterization of nanocellulose-reinforced nanocomposites: A review, *Adv. Colloid Interface Sci.*, 2023, 102961.
- 6 D. Klemm, B. Heublein, H. P. Fink and A. Bohn, Cellulose: fascinating biopolymer and sustainable raw material, *Angew. Chem., Int. Ed.*, 2005, **44**(22), 3358–3393.
- 7 K. Nagarajan, *et al.*, A comprehensive review on cellulose nanocrystals and cellulose nanofibers: Pretreatment, preparation, and characterization, *Polym. Compos.*, 2021, **42**(4), 1588–1630.
- 8 J. George and S. Sabapathi, Cellulose nanocrystals: synthesis, functional properties, and applications, *Nanotechnol., Sci. Appl.*, 2015, **8**, 45.
- 9 D. Dey, *et al.*, Physical, antifungal, and biodegradable properties of cellulose nanocrystals and chitosan nanoparticles for food packaging application, *Mater. Today: Proc.*, 2021, **38**, 860–869.
- 10 N. Nepomuceno, A. Seixas, E. Medeiros and T. Mélo, Evaluation of conductivity of nanostructured polyaniline/cellulose nanocrystals (PANI/CNC) obtained via in situ polymerization, *J. Solid State Chem.*, 2021, **302**, 122372.
- 11 J. C. Jackson, C. H. Camargos, V. T. Noronha, A. J. Paula, C. A. Rezende and A. F. Faria, Sustainable cellulose nanocrystals for improved antimicrobial properties of thin film composite membranes, *ACS Sustainable Chem. Eng.*, 2021, **9**(19), 6534–6540.
- 12 L. Qin, H. Gao, S. Xiong, Y. Jia and L. Ren, Preparation of collagen/cellulose nanocrystals composite films and their potential applications in corneal repair, *J. Mater. Sci.: Mater. Med.*, 2020, **31**, 1–11.
- 13 U. Mamudu, M. R. Hussin, J. H. Santos and R. C. Lim, Synthesis and characterisation of sulfated-nanocrystalline cellulose in epoxy coatings for corrosion protection of mild steel from sodium chloride solution, *Carbohydr. Polym. Technol. Appl.*, 2023, **5**, 100306.
- 14 N. F. S. M. Azani, M. M. Haafiz, A. Zahari, S. Poinsignon, N. Brosse and M. H. Hussin, Preparation and characterizations of oil palm fronds cellulose nanocrystal (OPF-CNC) as reinforcing filler in epoxy-Zn rich coating for mild steel corrosion protection, *Int. J. Biol. Macromol.*, 2020, **153**, 385–398.
- 15 N. F. S. M. Azani and M. H. Hussin, Comparison of cellulose nanocrystal (CNC) filler on chemical, mechanical, and corrosion properties of epoxy-Zn protective coatings for mild steel in 3.5% NaCl solution, *Cellulose*, 2021, **28**(10), 6523–6543.
- 16 X. Wang, K. Gao, E. B. Caldona, M. R. R. Ali, X. Zhang and Z. Zhang, Cellulose nanocrystals-reinforced waterborne epoxy coatings with enhanced corrosion resistance for steel, *Int. J. Biol. Macromol.*, 2024, **257**, 128755.
- 17 Y. He, Y. Boluk, J. Pan, A. Ahniyaz, T. Deltin and P. M. Claesson, Corrosion protective properties of cellulose nanocrystals reinforced waterborne acrylate-based composite coating, *Corros. Sci.*, 2019, **155**, 186–194.
- 18 S. Tamantini, S. Bergamasco, F. Zikeli, M. Humar, M. Cavalera and M. Romagnoli, Cellulose nano crystals (CNC) as additive for a bio-based waterborne acrylic wood coating: decay, artificial weathering, physical and chemical tests, *Nanomaterials*, 2023, **13**(3), 442.
- 19 M. N. Khan, C. M. Clarkson, M. Nuruddin, A. Sharif, E. Ahmad and J. P. Youngblood, Performance of advanced waterborne wood coatings reinforced with cellulose nanocrystals, *ACS Appl. Bio Mater.*, 2022, **5**(9), 4179–4190.
- 20 Z. Wang, B. Hu and G. Z. Chen, The role of 1-octyl-3-methylimidazolium hexafluorophosphate in anticorrosion coating formula development, *J. Saudi Chem. Soc.*, 2022, **26**(3), 101446.
- 21 Z. Wang, B. Hu, H. Yu and G. Z. Chen, Synergistic Effects of 1-Octyl-3-Methylimidazolium Hexafluorophosphate and Cellulose Nanocrystals on Improving Polyacrylate Waterborne Anti-Corrosion Coatings, *Polymers*, 2023, **15**(4), 810.



22 Y. Kong, *Encapsulation of Room Temperature Ionic Liquids by Miniemulsion Polymerisation for Application in Low-Emitting Latex Coating*, University of Nottingham, 2016.

23 T. Sakai, K. Kamogawa, F. Harusawa, N. Momozawa, H. Sakai and M. Abe, Direct observation of flocculation/coalescence of metastable oil droplets in surfactant-free oil/water emulsion by freeze-fracture electron microscopy, *Langmuir*, 2001, **17**(2), 255–259.

24 T. M. Ho, A. Razzaghi, A. Ramachandran and K. S. Mikkonen, Emulsion characterization via microfluidic devices: A review on interfacial tension and stability to coalescence, *Adv. Colloid Interface Sci.*, 2022, **299**, 102541.

25 Z. Hu, S. Ballinger, R. Pelton and E. D. Cranston, Surfactant-enhanced cellulose nanocrystal Pickering emulsions, *J. Colloid Interface Sci.*, 2015, **439**, 139–148.

26 K. Reinheimer, M. Gross and M. Wilhelm, Fourier transform rheology as a universal non-linear mechanical characterization of droplet size and interfacial tension of dilute monodisperse emulsions, *J. Colloid Interface Sci.*, 2011, **360**(2), 818–825.

27 K. von Nessen, M. Karg and T. Hellweg, Thermoresponsive poly-(N-isopropylmethacrylamide) microgels: Tailoring particle size by interfacial tension control, *Polymer*, 2013, **54**(21), 5499–5510.

28 I. Kalashnikova, H. Bizot, B. Cathala and I. Capron, New Pickering emulsions stabilized by bacterial cellulose nanocrystals, *Langmuir*, 2011, **27**(12), 7471–7479.

29 I. Kalashnikova, H. Bizot, B. Cathala and I. Capron, Modulation of cellulose nanocrystals amphiphilic properties to stabilize oil/water interface, *Biomacromolecules*, 2011, **13**(1), 267–275.

30 I. Kalashnikova, H. Bizot, P. Bertoncini, B. Cathala and I. Capron, Cellulosic nanorods of various aspect ratios for oil in water Pickering emulsions, *Soft Matter*, 2013, **9**(3), 952–959.

31 I. Capron and B. Cathala, Surfactant-free high internal phase emulsions stabilized by cellulose nanocrystals, *Biomacromolecules*, 2013, **14**(2), 291–296.

32 S. Tasset, B. Cathala, H. Bizot and I. Capron, Versatile cellular foams derived from CNC-stabilized Pickering emulsions, *RSC Adv.*, 2014, **4**(2), 893–898.

33 B. P. Binks and T. S. Horozov, *Colloidal Particles at Liquid Interfaces*, Cambridge University Press, 2006.

34 R. I. Dekker, *et al.*, Is there a difference between surfactant-stabilised and Pickering emulsions?, *Soft Matter*, 2023, **19**(10), 1941–1951.

35 T. Okubo, Surface tension of structured colloidal suspensions of polystyrene and silica spheres at the air–water interface, *J. Colloid Interface Sci.*, 1995, **171**(1), 55–62.

36 N. Saleh, T. Sarbu, K. Sirk, G. V. Lowry, K. Matyjaszewski and R. D. Tilton, Oil-in-water emulsions stabilized by highly charged polyelectrolyte-grafted silica nanoparticles, *Langmuir*, 2005, **21**(22), 9873–9878.

37 N. Glaser, D. J. Adams, A. Böker and G. Krausch, Janus particles at liquid–liquid interfaces, *Langmuir*, 2006, **22**(12), 5227–5229.

38 S. Kutuzov, J. He, R. Tangirala, T. Emrick, T. Russell and A. Böker, On the kinetics of nanoparticle self-assembly at liquid/liquid interfaces, *Phys. Chem. Chem. Phys.*, 2007, **9**(48), 6351–6358.

39 J. Forth, P. Y. Kim, G. Xie, X. Liu, B. A. Helms and T. P. Russell, Building reconfigurable devices using complex liquid–fluid interfaces, *Adv. Mater.*, 2019, **31**(18), 1806370.

40 D. Sun, S. Kang, C. Liu, Q. Lu, L. Cui and B. Hu, Effect of zeta potential and particle size on the stability of SiO₂ nanospheres as carrier for ultrasound imaging contrast agents, *Int. J. Electrochem. Sci.*, 2016, **11**(10), 8520–8529.

41 D. J. Pochapski, C. Carvalho dos Santos, G. W. Leite, S. H. Pulcinelli and C. V. Santilli, Zeta potential and colloidal stability predictions for inorganic nanoparticle dispersions: Effects of experimental conditions and electrokinetic models on the interpretation of results, *Langmuir*, 2021, **37**(45), 13379–13389.

42 P. A. Lovell and F. J. Schork, Fundamentals of emulsion polymerization, *Biomacromolecules*, 2020, **21**(11), 4396–4441.

43 P. Jenkins and M. Snowden, Depletion flocculation in colloidal dispersions, *Adv. Colloid Interface Sci.*, 1996, **68**, 57–96.

44 A. Wittmar, D. Ruiz-Abad and M. Ulbricht, Dispersions of silica nanoparticles in ionic liquids investigated with advanced rheology, *J. Nanopart. Res.*, 2012, **14**(2), 651.

45 N. Bechthold and K. Landfester, Kinetics of miniemulsion polymerization as revealed by calorimetry, *Macromolecules*, 2000, **33**(13), 4682–4689.

46 A. Einstein, Über die von der molekularkinetischen Theorie der Wärme geforderte Bewegung von in ruhenden Flüssigkeiten suspendierten Teilchen, *Ann. Phys.*, 1905, **322**(8), 549–560.

47 S. Eve and J. Mohr, Effects of UV-irradiation on the thermo-mechanical properties of optical grade poly (methyl methacrylate), *Appl. Surf. Sci.*, 2010, **256**(9), 2927–2933.

48 B. Schneider, J. Štokr, P. Schmidt, M. Mihailov, S. Dirlíkov and N. Peeva, Stretching and deformation vibrations of CH₂, C (CH₃) and O (CH₃) groups of poly (methyl methacrylate), *Polymer*, 1979, **20**(6), 705–712.

49 G. Duan, C. Zhang, A. Li, X. Yang, L. Lu and X. Wang, Preparation and characterization of mesoporous zirconia made by using a poly (methyl methacrylate) template, *Nanoscale Res. Lett.*, 2008, **3**(3), 118.

50 S. Ramesh, K. H. Leen, K. Kumutha and A. Arof, FTIR studies of PVC/PMMA blend based polymer electrolytes, *Spectrochim. Acta, Part A*, 2007, **66**(4–5), 1237–1242.

51 M. Schwanninger, J. Rodrigues, H. Pereira and B. Hinterstoisser, Effects of short-time vibratory ball milling on the shape of FT-IR spectra of wood and cellulose, *Vib. Spectrosc.*, 2004, **36**(1), 23–40.

52 J. Lamaming, R. Hashim, O. Sulaiman, C. P. Leh, T. Sugimoto and N. A. Nordin, Cellulose nanocrystals isolated from oil palm trunk, *Carbohydr. Polym.*, 2015, **127**, 202–208.



53 A. Mandal and D. Chakrabarty, Isolation of nanocellulose from waste sugarcane bagasse (SCB) and its characterization, *Carbohydr. Polym.*, 2011, **86**(3), 1291–1299.

54 W. Wulandari, A. Rochliadi and I. Arcana, Nanocellulose prepared by acid hydrolysis of isolated cellulose from sugarcane bagasse, in *IOP Conference Series: Materials Science and Engineering*, IOP Publishing, 2016, vol. 107, supp. 1, p. 012045.

55 A. Alemdar and M. Sain, Isolation and characterization of nanofibers from agricultural residues–Wheat straw and soy hulls, *Bioresour. Technol.*, 2008, **99**(6), 1664–1671.

56 Y. Liu, *et al.*, Modified ammonium persulfate oxidations for efficient preparation of carboxylated cellulose nanocrystals, *Carbohydr. Polym.*, 2020, **229**, 115572.

57 B. Abderrahim, E. Abderrahman, A. Mohamed, T. Fatima, T. Abdesselam and O. Krim, Kinetic thermal degradation of cellulose, polybutylene succinate and a green composite: comparative study, *World J. Environ. Eng.*, 2015, **3**(4), 95–110.

58 C. Campano, P. Lopez-Exposito, L. Gonzalez-Aguilera, Á. Blanco and C. Negro, In-depth characterization of the aggregation state of cellulose nanocrystals through analysis of transmission electron microscopy images, *Carbohydr. Polym.*, 2021, **254**, 117271.

59 N. Anton and T. F. Vandamme, Nano-emulsions and micro-emulsions: clarifications of the critical differences, *Pharm. Res.*, 2011, **28**(5), 978–985.

60 Q. Yuan and R. A. Williams, CO-stabilisation mechanisms of nanoparticles and surfactants in Pickering Emulsions produced by membrane emulsification, *J. Membr. Sci.*, 2016, **497**, 221–228.

61 K. Guo, P. Wei, Y. Xie and X. Huang, Smart ultra-stable foams stabilized using cellulose nanocrystal (CNC) gels via noncovalent bonding, *Chem. Commun.*, 2022, **58**(30), 4723–4726.

62 J. Hu, *et al.*, Carboxylation of cellulose nanocrystals for reinforcing and toughing rubber through dual cross-linking networks, *ACS Appl. Polym. Mater.*, 2021, **3**(12), 6120–6129.

63 T. Cao and M. Elimelech, Colloidal stability of cellulose nanocrystals in aqueous solutions containing monovalent, divalent, and trivalent inorganic salts, *J. Colloid Interface Sci.*, 2021, **584**, 456–463.

