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## Influence of ethylene thermal decomposition on carbon nanotube growth: insights from a two-zone reactor study†

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Gas-phase decomposition of the carbon precursor is a critical yet not fully understood step in the catalytic chemical vapor deposition growth of carbon nanotubes (CNTs). Here, we present a systematic investigation of how the thermal decomposition of C<sub>2</sub>H<sub>4</sub> influences CNT growth. Using a custom-designed two-zone reactor with independently controlled preheating (decomposition) and CNT growth zones, we decoupled the effects of gas-phase decomposition from the growth temperature. Standard synthesis conditions were first established using Bayesian optimization, with the CNT growth zone temperature ( $T_g$ ) fixed at 700 °C. CNTs were then synthesized at preheating zone temperatures ( $T_p$ ) of 500, 600, 700, 800, and 900 °C while maintaining  $T_g$  at 700 °C under standard conditions. Carbon yield and  $I_G/I_D$  remained stable at 500, 600, and 700 °C but showed significant changes at 800 and 900 °C. To elucidate these variations, we analyzed the gas-phase composition at various temperatures using micro-GC. Significant changes in the gas-phase composition were observed above 700 °C, correlating with changes in carbon yield and crystallinity. Based on these results, we propose a mechanism by which differences in gas-phase chemistry lead to changes in the carbon yield and the  $I_G/I_D$  ratio. This study provides valuable insight into the role of gas-phase decomposition in CNT growth and highlights the potential of tuning gas-phase chemistry for controlled growth of CNTs.

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## 1. Introduction

Carbon nanotubes (CNTs) have received considerable attention across various applications due to their outstanding mechanical,<sup>1</sup> electrical,<sup>2</sup> and thermal properties,<sup>3</sup> combined with low density. However, in the early stages of CNT development, their high cost and limited production capacity restricted their practical use. In recent years, significant progress has been made in the mass production of CNTs, particularly multi-walled CNTs (MWCNTs), *via* catalytic chemical vapor deposition (CCVD) in fluidized bed reactors,<sup>4–8</sup> using impregnated catalysts such as Fe/Al<sub>2</sub>O<sub>3</sub>,<sup>9</sup> Co/MgO,<sup>10</sup> and Co–Mo/Al<sub>2</sub>O<sub>3</sub>.<sup>11</sup> Furthermore, the CNT market has grown rapidly following the

realization that CNTs can serve as more effective conductive additives than conventional conductive additives such as carbon black.<sup>12</sup> This improvement is mainly attributed to their superior ability to form an electrical percolation network.

The next major challenge in expanding the CNT market is achieving mass production of CNTs with high crystallinity. This focus is driven by the well-established correlation between CNT crystallinity and their key properties, as well as overall application performance. For instance, higher crystallinity enhances thermal stability,<sup>13</sup> increases the Young's modulus of individual CNTs,<sup>14</sup> improves the electrical conductivity of CNT fibers,<sup>15</sup> bucky paper,<sup>16</sup> and films,<sup>17</sup> and elevates the tensile strength of CNT fibers.<sup>15,18</sup> While CNT crystallinity is known to increase with growth temperature,<sup>19</sup> this approach presents trade-offs. Increasing the growth temperature often leads to decreased yield, likely due to shortened catalyst lifetime and nucleation efficiency.<sup>20</sup> In addition, higher growth temperature requires higher energy consumption. Therefore, developing alternative strategies to improve CNT crystallinity without raising the growth temperature is essential for scalable and energy-efficient production.

To enhance CNT crystallinity, extensive research has been conducted on how various synthesis parameters influence their crystalline structure. Lee *et al.*<sup>21</sup> reported variations in

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CNT crystallinity depending on the type of metal species, while Kimura *et al.*<sup>22</sup> demonstrated changes based on the type of carbon precursor. Abbaslou *et al.*<sup>23</sup> investigated how the concentration of the carbon precursor affects crystallinity, and Moon *et al.*<sup>24</sup> showed that using Ar as an additional carrier gas can further improve CNT crystallinity. In addition, Shen *et al.*<sup>25</sup> introduced a defect-healing strategy *via* thermal treatment in the presence of air, which effectively improved the crystallinity of CNTs. Despite these efforts, no universally effective method—aside from increasing the growth temperature—has been established across general CNT synthesis systems. As a result, the scalable production of high-crystallinity CNTs still remains a significant challenge.

Here, we investigated how the thermal decomposition of  $C_2H_4$ , used as a carbon precursor, influences the crystallinity and carbon yield of CNTs grown on a Co/MgO catalyst prepared *via* wet impregnation. In a typical CVD reactor for CNT growth, it is difficult to have different temperatures for the gas-phase decomposition of the carbon precursor and the catalytic growth of CNTs.<sup>26–29</sup> To decouple these temperatures, we employed a customized two-zone reactor consisting of a preheating zone and a CNT growth zone, allowing independent temperature control in each zone. Before assessing the impact of decomposition by varying the preheating zone temperature ( $T_p$ ), we optimized the CNT growth conditions for carbon yield using Bayesian optimization (BO), maintaining the CNT growth temperature ( $T_g$ ) at 700 °C to establish standard synthesis conditions. We then synthesized CNTs at  $T_p$  values of 500, 600, 700, 800, and 900 °C under these standard conditions and measured the carbon yield and  $I_G/I_D$  ratio. The carbon yield and  $I_G/I_D$  ratio remained consistent at  $T_p$  of 500, 600, and 700 °C, but varied at 800 and 900 °C. To elucidate these variations, we analyzed the composition and concentration of the gases formed by  $C_2H_4$  decomposition in the preheating zone using micro-gas chromatography (micro-GC). Based on the gas analysis data, we propose a mechanism by

which differences in gas chemistry lead to changes in the carbon yield and the  $I_G/I_D$  ratio.

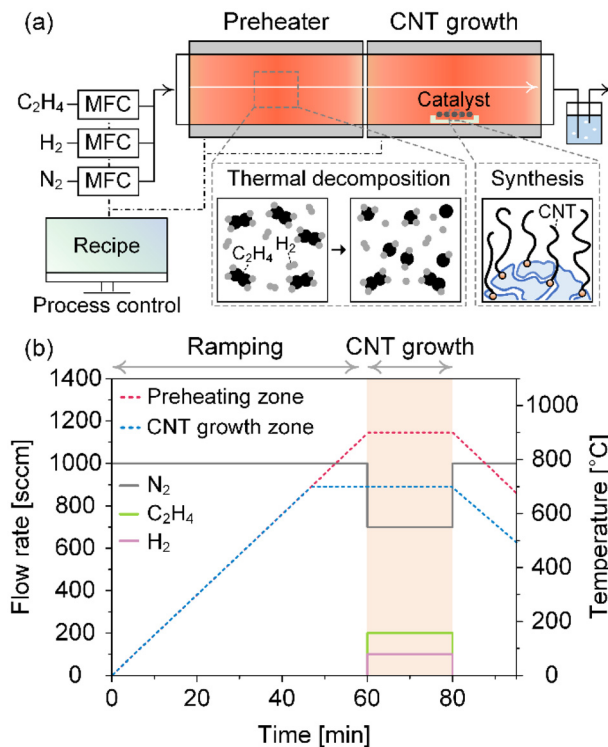
## 2. Experimental

### 2.1. Catalyst preparation

The Co/MgO catalysts were prepared *via* the wet impregnation method. First, porous MgO powder (98%, average particle diameter  $\sim 93$   $\mu\text{m}$ , measured by particle size analysis, Thermo Fisher Scientific) was stirred in an aqueous solution at room temperature for one hour. To achieve a 25 wt% cobalt loading, the necessary amount of cobalt nitrate hexahydrate (98.0%, Matsugaki Chemical Industries) was then added to the suspension, followed by stirring for one additional hour at room temperature. The resulting mixture was dried at 170 °C for 20 hours under continuous stirring. The dried sample was subsequently ground and calcined in air at 700 °C for two hours.

### 2.2. CNT synthesis

CNT synthesis was carried out in a customized two-zone reactor consisting of a preheating zone and a CNT growth zone, allowing independent temperature control in each (Fig. 1a). We used a quartz tube with an inner diameter of 5.5 cm and a length of 1.3 m. For each run, we utilized 0.02 g of Co/MgO catalyst to synthesize CNTs. We used three types of



**Fig. 1** (a) Scheme of the customized two-zone reactor consisting of a preheating zone and a CNT growth zone, operating under a computer-controlled, programmed recipe for CNT growth. (b) An example of a decoupled growth recipe at a  $T_p$  of 900 °C and a  $T_g$  of 700 °C.



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gases:  $C_2H_4$  as a carbon precursor,  $H_2$ , and  $N_2$ . The growth process was run by a programmed recipe using a computer to precisely control experimental conditions and eliminate human error (Fig. 1b and Fig. S1†). We synthesized CNTs at  $T_p$  of 500, 600, 700, 800, and 900 °C, maintaining  $T_g$  at 700 °C for 20 minutes. The ramping rate in both zones was approximately 15 °C  $min^{-1}$ . The total flow rate,  $C_2H_4$  concentration, and  $H_2$  concentration were determined as recommended by BO, and the  $N_2$  flow rate was decided by subtracting the  $C_2H_4$  and  $H_2$  flow rates from the total flow rate.

After synthesis, we measured the carbon yield, which is defined as the mass of carbon produced during CNT synthesis per unit mass of catalyst, and measured the mass of the final product. The carbon yield can be expressed using the following equation:

$$\text{Carbon yield [\%]} = \frac{M_f - M_{\text{cat}}}{M_{\text{cat}}} \times 100$$

where  $M_f$  is the mass of the final product (containing the carbon and the Co/MgO catalyst) and  $M_{\text{cat}}$  is the mass of the Co/MgO catalyst loaded prior to CNT synthesis.

### 2.3. Bayesian optimization

We used BoTorch as the library.<sup>30</sup> We employed the Matern 5/2 kernel function for the Gaussian process and used Expected Improvement (EI) as the acquisition function. The BO process began by building an initial experimental database, which was then fed into the BO algorithm to generate a new set of input variables for further investigation. Next, the experiment was performed with the newly generated variables, and the resulting data were added to the database. This iterative cycle continued until the variables were optimized.

### 2.4. Characterization

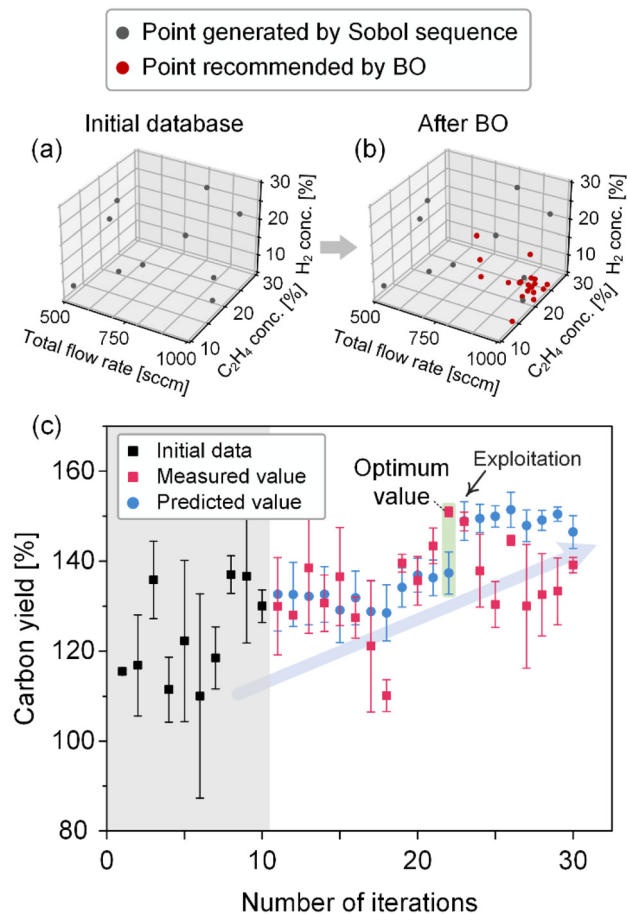
We characterized the as-synthesized CNT samples using several analytical techniques, including scanning electron microscopy (SEM), transmission electron microscopy (TEM), Raman spectroscopy, and thermogravimetric analysis (TGA). The SEM analysis was conducted using a SUPRA 25 microscope from Carl Zeiss AG in Oberkochen, Germany. For TEM analysis, we employed an FEI-Titan Cubed 60–300 microscope equipped with a Cs-corrector and monochromator, operating at 80 kV. To assess the ratio of the intensity of the D-band ( $I_D$ ) to the G-band ( $I_G$ ) in the Raman spectra, we utilized an NS240-F Raman spectrometer from Nanoscope Systems in Daejeon, Republic of Korea. The laser excitation wavelength was set to 532 nm. Finally, we investigated the purity of the CNT samples using TGA with a Discovery TGA 55 instrument from TA Instruments in New Castle, DE, USA, which was conducted in air from 30 °C to 950 °C with a heating rate of 5 °C  $min^{-1}$ .

## 3. Results and discussion

### 3.1. Maximizing carbon yield using Bayesian optimization

Before assessing the impact of decomposition of  $C_2H_4$  by varying  $T_p$ , it was necessary to first establish standard CNT

growth conditions. For this, we optimized the CNT growth conditions to maximize carbon yield using BO. We selected three input variables in the CNT synthesis process: total flow rate, percentage of  $C_2H_4$  in the total flow rate, and percentage of  $H_2$  in the total flow rate. We designated the ranges for these variables: 500–1000 sccm, 10.0–30.0%, and 6.0–30.0%. These



**Fig. 2** (a) Three-dimensional plot illustrating the points generated by the Sobol sequence for the initial database (gray) and (b) the points recommended by BO (red) within the input variable space. (c) The predicted and measured carbon yields for each iteration, along with the initial database.

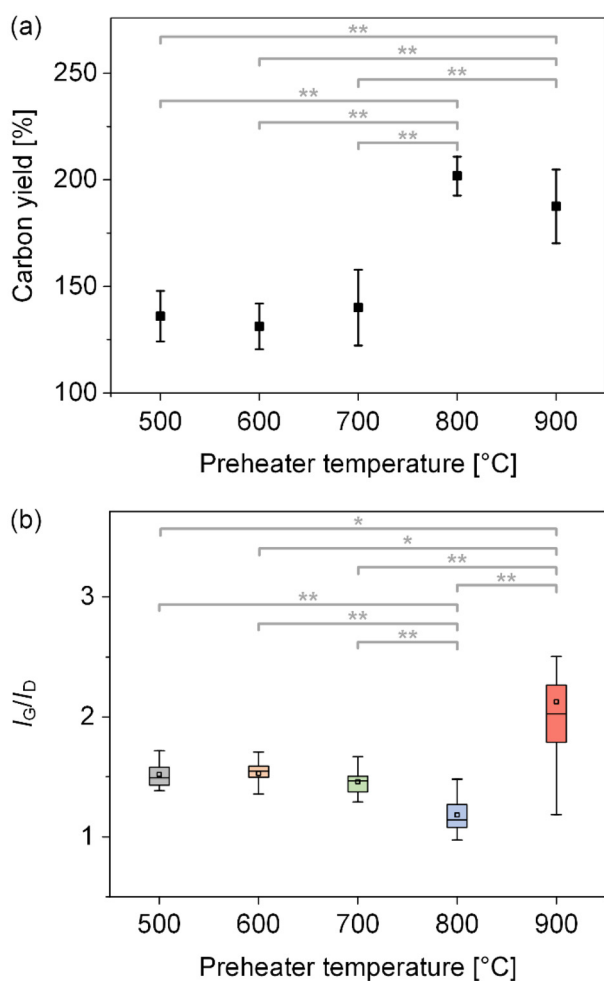
**Table 1** Initial database built using the Sobol sequence to initiate BO

Number	Total flow rate [sccm]	$C_2H_4$ flow rate [sccm]	$H_2$ flow rate [sccm]	$N_2$ flow rate [sccm]	Carbon yield [%]
1	910	220	153	537	115.5 ± 0.7
2	597	95	143	359	116.8 ± 15.9
3	734	208	87	439	135.8 ± 12.2
4	797	92	153	552	111.4 ± 10.2
5	850	221	247	382	122.3 ± 25.4
6	661	92	83	486	110.0 ± 32.1
7	540	116	127	297	118.5 ± 9.8
8	978	182	67	729	137.0 ± 6.0
9	950	273	205	472	136.7 ± 21.0
10	514	57	41	416	130.0 ± 5.2



**Table 2** Recommended points (sets of input variables) by BO, their predicted values, and experimentally measured carbon yields

Iteration number	Total flow rate [sccm]	C <sub>2</sub> H <sub>4</sub> flow rate [sccm]	H <sub>2</sub> flow rate [sccm]	N <sub>2</sub> flow rate [sccm]	Predicted value [%]	Carbon yield [%]
1	867	195	52	620	132.6 ± 8.2	123.0 ± 15.3
2	1000	138	60	802	132.6 ± 7.1	128.0 ± 1.3
3	1000	233	62	705	132.1 ± 6.3	138.5 ± 20.5
4	1000	245	64	691	132.6 ± 6.2	130.7 ± 8.9
5	717	177	55	485	129.1 ± 7.2	136.6 ± 15.4
6	778	161	58	559	131.8 ± 6.0	127.4 ± 6.5
7	627	188	53	386	128.8 ± 6.8	121.1 ± 20.6
8	858	258	64	536	128.5 ± 6.2	110.1 ± 4.9
9	1000	203	60	737	134.2 ± 4.4	139.6 ± 2.8
10	955	193	57	705	136.9 ± 3.8	135.6 ± 7.7
11	1000	202	81	717	136.3 ± 4.0	143.4 ± 5.6
12	1000	196	103	701	137.4 ± 4.7	150.9 ± 1.5
13	1000	185	118	697	148.9 ± 4.3	148.8 ± 3.0
14	1000	204	116	680	149.5 ± 3.1	137.9 ± 11.4
15	1000	183	103	714	149.9 ± 2.4	130.4 ± 7.3
16	1000	193	130	677	151.4 ± 3.9	144.6 ± 1.5
17	956	187	102	667	147.9 ± 3.5	123.0 ± 19.5
18	1000	205	103	692	149.1 ± 2.3	132.5 ± 12.9
19	1000	193	104	703	150.4 ± 1.6	133.3 ± 10.5
20	954	184	102	668	146.5 ± 3.6	139.2 ± 2.4

**Fig. 3** (a) Carbon yield and (b)  $I_G/I_D$  ratio as a function of  $T_p$ . \*  $p$ -value < 0.05, \*\*  $p$ -value < 0.01.

ranges were experimentally determined based on the physical limitations of the system. The flow rate of N<sub>2</sub> was decided by subtracting the flow rates of C<sub>2</sub>H<sub>4</sub> and H<sub>2</sub> from the total flow rate. We fixed the temperature of the CNT growth zone at 700 °C.

To initiate BO, we constructed an initial database using the Sobol sequence, a quasi-random number generator that generates evenly distributed points within the input variable space, thereby reducing the clustering often seen with purely random sampling.<sup>31</sup> Having an evenly distributed initial set of points helps build a robust surrogate model, guides the acquisition function more effectively, and ultimately leads to a more efficient and successful optimization process. Therefore, we generated 10 points within the input space using the Sobol sequence (Fig. 2a) and conducted two repeated experiments at each point. The maximum carbon yield observed in the initial database was 137.0% (Table 1).

After building the initial database, we performed 20 iterations with two repeated experiments per iteration (Fig. 2c and Table 2). Through BO, we successfully optimized the CNT growth conditions to increase the carbon yield, ultimately establishing the standard conditions. We achieved a maximum carbon yield of 150.9% at the 12th iteration and noted a general upward trend in carbon yield as the iterations progressed. During the optimization, we observed the acquisition function's struggle in balancing exploration (sampling points with high uncertainty) and exploitation (sampling points with high predicted values) to find the global optimum. In the 11th iteration, the measured value significantly exceeded the model's prediction. This unexpected result led the acquisition function to recommend sampling points near the input variable region where the measured value had sharply risen. This behavior exemplified the function's exploitation strategy.



### 3.2. Influence of carbon precursor decomposition on CNT growth

To identify the influence of carbon precursor decomposition on CNT growth, we synthesized CNTs at different  $T_p$  of 500, 600, 700, 800, and 900 °C under the standard synthesis conditions and measured the carbon yield and  $I_G/I_D$  ratio (Fig. 3, Table S1, and Fig. S2†). The  $I_G/I_D$  ratio represents the intensity ratio of the G peak (around 1580  $\text{cm}^{-1}$ ) to the D peak (around 1350  $\text{cm}^{-1}$ ) in the Raman spectra. Specifically, the G peak indicates the ordered  $\text{sp}^2$  networks of carbon atoms in CNTs, whereas the D peak arises from structural defects and disorder. Consequently, a higher  $I_G/I_D$  ratio signifies a greater proportion of graphitic (ordered) structures relative to non-graphitic (disordered or defective) structures in the CNTs. To obtain reliable data for carbon yield and  $I_G/I_D$  ratio, we performed five repeated CNT syntheses at each  $T_p$ . For each CNT sample synthesized in a single run, we collected samples at three different locations and measured the Raman spectrum at five

different spots per sample. Consequently, we synthesized a total of 25 CNT samples and obtained 15 Raman spectra per sample, yielding a total of 375 Raman spectra.

We investigated how the carbon yield and  $I_G/I_D$  ratio varied with  $T_p$  using Student's  $t$ -test. Student's  $t$ -test is a statistical method commonly employed to determine whether the difference between the means of two groups is statistically significant. This process begins by formulating a null hypothesis (assuming no difference between group means) and an alternative hypothesis (assuming a significant difference). A  $t$ -statistic is calculated based on the means, standard deviations, and sample sizes of the two groups. Using this  $t$ -statistic and the corresponding degrees of freedom, a  $p$ -value is obtained from the  $t$ -distribution. If the  $p$ -value is below a pre-defined significance level (usually 0.05), the null hypothesis is rejected, indicating a statistically significant difference between the groups; otherwise, the null hypothesis is retained, suggesting no significant difference. In this study, we performed a two-sided  $t$ -test with a significance level of 0.05.

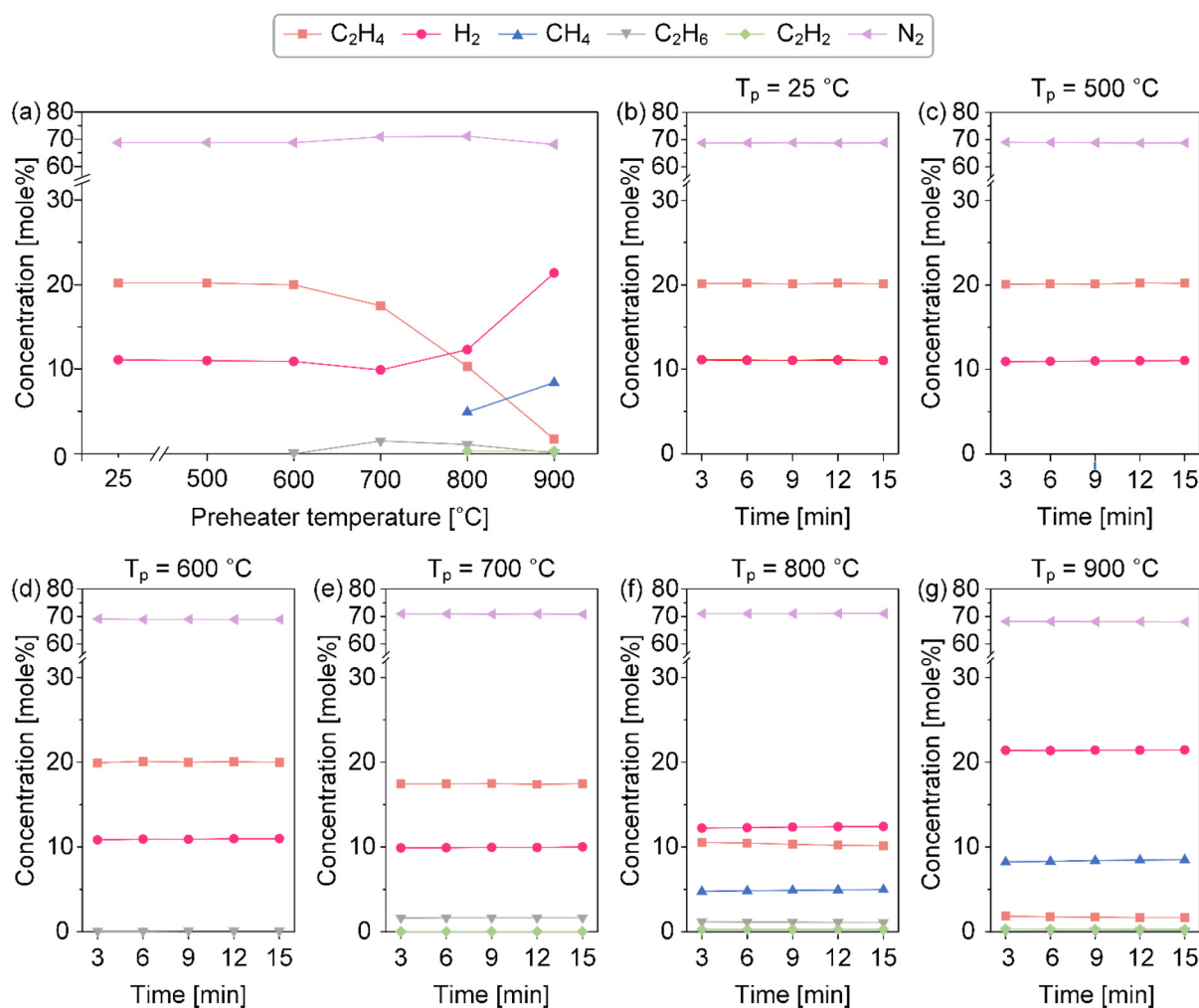


Fig. 4 (a) Gas evolution profiles resulting from feedstock gas decomposition at various  $T_p$  values, and (b–g) gas evolution profiles over time at  $T_p$  = 25, 500, 600, 700, 800, and 900 °C.



We first compared the mean carbon yield and found significant differences between the following  $T_p$  pairs: 500 and 800 °C, 500 and 900 °C, 600 and 800 °C, 600 and 900 °C, 700 and 800 °C, and 700 and 900 °C, with each  $p$ -value less than 0.01 (Fig. 3a). That is, the carbon yield remained consistent at 500, 600, and 700 °C, then increased at 800 °C, and showed no significant difference between 800 and 900 °C. Consistent with this trend, TGA of samples synthesized at 700, 800, and 900 °C (Fig. S3†) reveals a final weight of about 45.0 wt% at 700 °C, and approximately 32 wt% at both 800 °C and 900 °C—further implying that the carbon yields at 800 °C and 900 °C are comparable and higher than at 700 °C. Additionally, we observed that the residual mass after TGA correlated closely with the carbon yield. Next, we compared the mean  $I_G/I_D$  ratio and found significant differences between the following  $T_p$  pairs: 500 and 800 °C, 500 and 900 °C, 600 and 800 °C, 600 and 900 °C, 700 and 800 °C, 700 and 900 °C, and 800 and 900 °C. This indicates that the  $I_G/I_D$  ratio remained consistent at 500, 600, and 700 °C, decreased at 800 °C, and rose sharply at 900 °C.

Generally, it is known that the  $I_G/I_D$  ratio increases with higher  $T_g$ . An interesting point in our synthesis results is that the  $I_G/I_D$  ratio varied with  $T_p$ , despite maintaining  $T_g$  at 700 °C. Although there are a few studies on the synthesis of CNTs using a preheater, this behavior has not been reported in previous studies. Meshot *et al.* used a cold-walled CNT forest synthesis system and observed that the  $I_G/I_D$  ratio decreased linearly with increasing  $T_p$  (900–1120 °C) when the growth substrate was held at 825 °C.<sup>32</sup> In contrast, Nessim *et al.* observed a more crystalline structure in MWCNTs (as seen in TEM images) with increasing  $T_p$  (730–770 °C) under a hot-walled synthesis system, with  $T_g$  set at 475 °C; however, they did not

provide the  $I_G/I_D$  ratio.<sup>33</sup> Similarly, Sugime *et al.* presented Raman spectra of CNT forests synthesized at  $T_p$  of 800, 900, and 1000 °C, maintaining the temperature of the growth substrate at 800 °C. Although the  $I_G/I_D$  ratio was not explicitly reported, no significant differences were observed in the intensities of the G and D peaks.<sup>34</sup>

To understand why the  $I_G/I_D$  ratio varied with  $T_p$ , we analyzed the composition and concentration of gases passing through the preheater at 25, 500, 600, 700, 800, and 900 °C using a micro-GC (Fig. 4a and Table S2†). Micro-GC, which is a gas chromatography equipped with packed columns and a thermal conductivity detector, enables real-time quantitative analysis of gas components. The detectable gas species included  $C_2H_4$ ,  $H_2$ ,  $CH_4$ ,  $C_2H_6$ ,  $C_2H_2$ ,  $N_2$ ,  $CO$ , and  $CO_2$ ; however,  $CO$  and  $CO_2$  were not observed under our synthesis processes. The temperature of the gaseous products may drop from the heater to the micro-GC instrument. The volume of each gas could change depending on the temperature, but all the gaseous compounds were passed through the same temperature gradient. Therefore, the relative distribution of gaseous compounds from the heater to the micro-GC instrument should not be changed. The GC analysis of gaseous compounds produced in the high-temperature reactors is a widely accepted method. As  $T_p$  increased, we observed tar compounds formed by the decomposition of carbon precursors on the quartz tube wall, which can block the reactor outlet or reduce the analytical sensitivity of the micro-GC if introduced into the system. To prevent tar from entering the micro-GC and ensure accurate gas analysis, we installed a cold trap and a filter upstream of the GC inlet to capture condensable tar components and particulates (Fig. S4†).

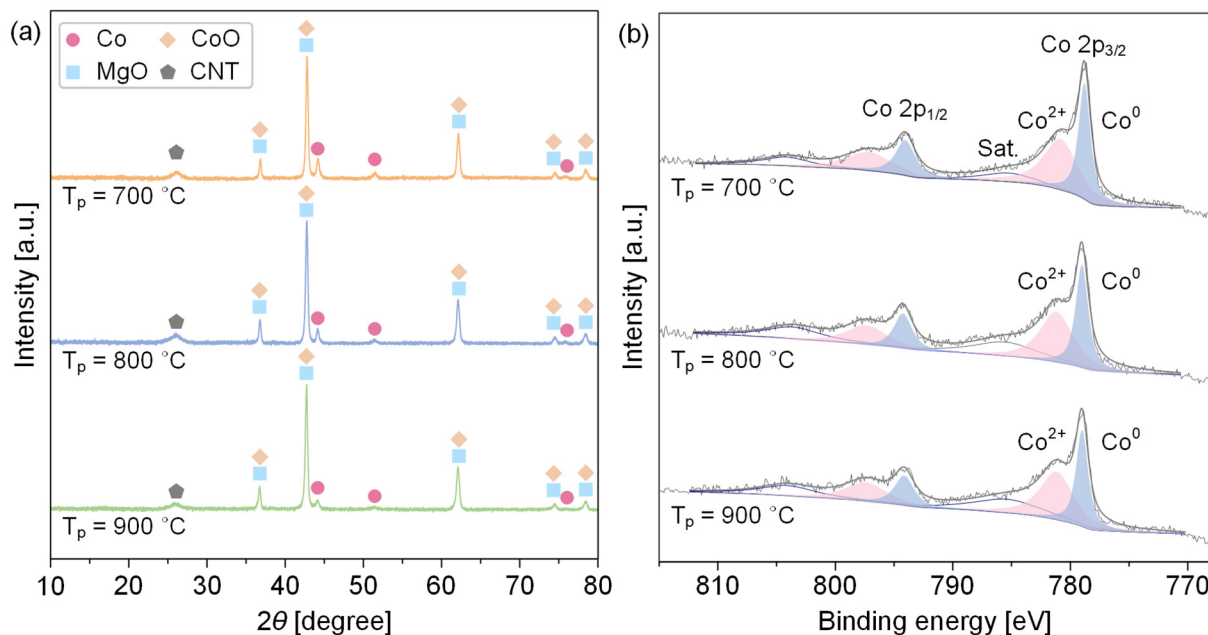


Fig. 5 (a) XRD patterns, and (b) Co  $2p_{1/2}$  and Co  $2p_{3/2}$  XPS spectra of the catalyst particles from each CNT sample synthesized at  $T_p$  of 700, 800, and 900 °C.



The decomposition of carbon precursors rarely occurred at  $T_p$  of 25, 500, and 600 °C, and slightly appeared at 700 °C. At 800 and 900 °C, active decomposition was observed, along with the formation of  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ , and  $\text{C}_2\text{H}_2$ , and an increasing concentration of  $\text{H}_2$ . Notably, a significant amount of  $\text{CH}_4$  was generated, and the concentration of  $\text{H}_2$  leapt at 900 °C. Because the ethylene decomposition is an endothermic reaction, higher temperatures promote the ethylene decomposition reaction further, with more formation of decomposed products. It is also known that ethylene decomposition and the simultaneous formation of other compounds start from slightly higher than 700 °C without a catalyst.<sup>35</sup> Additionally, we monitored the composition and concentration of these gases over time at each  $T_p$  (Fig. 4b–g) and found that they remained nearly constant. Therefore, we concluded reasonably that the decomposition of the carbon precursor occurred under steady-state conditions.

One notable result was that the concentration of  $\text{H}_2$  increased with  $T_p$ . Since reducing the catalyst from its oxide form to a crystalline metallic state is a critical step in CNT synthesis—and the metallic state is considered the active site for CNT growth<sup>36</sup>—we hypothesized that the intensity of Co peaks in the XRD patterns would increase with  $T_p$ , potentially reaching a maximum at 900 °C. To investigate the effect of  $\text{H}_2$  concentration on catalyst reduction, we performed XRD on the catalyst particles in each CNT sample synthesized at  $T_p$  of 700, 800, and 900 °C (Fig. 5a). The XRD patterns revealed the same set of characteristic peaks for Co (00-015-0806), CoO (01-075-0418), MgO (01-074-1225), and CNT (01-075-1621), with metallic cobalt reflections typically appearing at 44.2°, 51.5°, and 75.9°; CoO peaks around 36.5°, 42.4°, 61.5°, and 73°; MgO peaks near 42.9°, 62.3°, and 74.7°; and a broad peak from CNTs around 26°. Surprisingly, however, there was no significant difference in the relative intensity of these Co peaks among the three samples, indicating that—even at elevated  $T_p$ —the degree of Co reduction in the catalyst was broadly similar under all conditions.

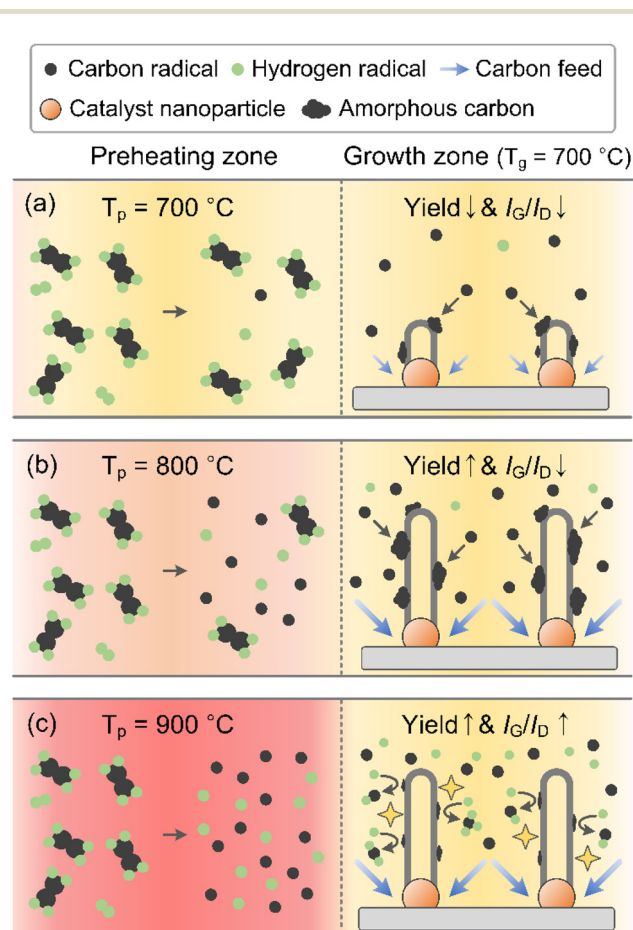
To explore potential subtle differences in Co peak intensity in the XRD spectra, we conducted XPS analysis of the Co  $2p_{1/2}$  and Co  $2p_{3/2}$  regions on the catalyst particles from each CNT sample synthesized at  $T_p = 700, 800,$  and  $900$  °C (Fig. 5b). XPS peak deconvolution revealed both metallic Co ( $\text{Co}^0$ ) and  $\text{Co}^{2+}$ , indicating almost no difference in their relative proportions among the three samples. Specifically, the CNTs synthesized at 700 °C contained 44.5%  $\text{Co}^0$  and 55.5%  $\text{Co}^{2+}$ , those at 800 °C contained 44.4%  $\text{Co}^0$  and 55.6%  $\text{Co}^{2+}$ , and those at 900 °C contained 44.2%  $\text{Co}^0$  and 55.8%  $\text{Co}^{2+}$ . Based on the combined XRD and XPS data, we rejected our initial hypothesis and reasonably concluded that the catalyst contributed equally to CNT growth at  $T_p = 700, 800,$  and  $900$  °C.

### 3.3. Model explaining CNT synthesis dependence on $T_p$

To explain the observed variations in the  $I_G/I_D$  ratio and carbon yield as  $T_p$  increases, except for the catalyst's influence, we needed to consider the gas-phase chemistry of the carbon and hydrogen radicals generated by the decomposition of the

carbon precursor. Although many researchers emphasize non-dissociative hydrocarbon molecules as the primary precursor for CNT growth, carbon and hydrogen radicals are rarely included in conventional growth models. Here, we propose a model that incorporates these radicals. As discussed earlier, we postulated that the decomposition of the carbon precursor occurs under steady-state conditions and the catalysts contribute equally to CNT growth at  $T_p = 700, 800,$  and  $900$  °C.

Our gas analysis (Fig. 4) revealed that the extent of carbon precursor decomposition depends strongly on  $T_p$ , with notable consequences for CNT growth. Typically, CNT formation follows a vapor–liquid–solid mechanism, where vapor-phase carbon precursors dissolve into liquid-phase metallic catalyst nanoparticles and then precipitate as solid CNTs.<sup>37</sup> As a result, the rate at which carbon precursors feed the catalyst must be carefully balanced with the precipitation rate. If this feed rate is too low, the carbon yield remains limited; if it is too high, it can encapsulate the catalyst particles in carbon. Thus, an optimal feed rate of carbon precursors is critical for efficient CNT growth.



**Fig. 6** Schematic illustration of the proposed CNT growth model at different  $T_p$ : (a) 700, (b) 800, and (c) 900 °C. Carbon and hydrogen radicals form in the preheating zone and migrate to the growth zone ( $T_g = 700$  °C), where they interact with catalyst nanoparticles. As  $T_p$  increases, higher concentrations of radicals influence both the carbon yield and the  $I_G/I_D$  ratio, leading to the varying growth outcomes depicted.



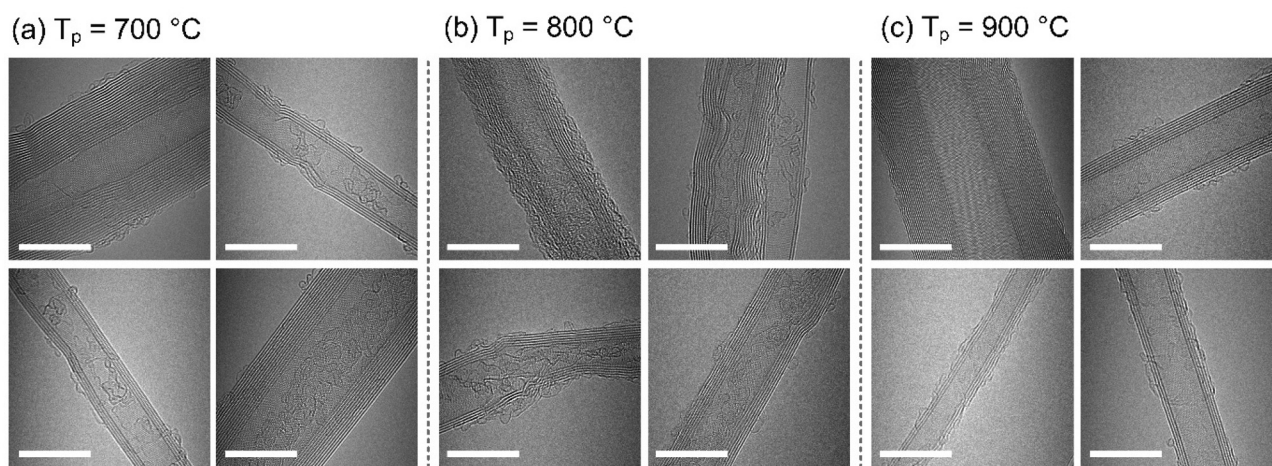


Fig. 7 TEM images of the synthesized CNTs at  $T_p$  of (a) 700, (b) 800, and (c) 900 °C (scale bars: 10 nm).

At 700 °C, only about 12% of the injected  $C_2H_4$  decomposed, generating a low concentration of carbon radicals and limiting the overall carbon yield (Fig. 6a). Because the feed rate is insufficient relative to the catalyst's precipitation rate, the overall CNT growth remains restricted. At 800 °C, however, around 50% of  $C_2H_4$  decomposed, significantly increasing both carbon and hydrogen radical concentrations and boosting the carbon yield (Fig. 6b). This improvement can be attributed to an increased precipitation rate enabled by the sufficient supply of carbon radicals to the catalyst. The excess carbon radicals, however, accumulate on CNT surfaces to form amorphous carbon, as a relatively lower  $I_G/I_D$  ratio is shown at preheating of 800 °C compared to those at 700 and 900 °C (Fig. 3b). Meanwhile, most hydrogen radicals are consumed in forming other hydrocarbons—as evidenced by only a slight (about 2%) rise in  $H_2$  concentration—thus leaving fewer hydrogen radicals to etch away the amorphous carbon at 800 °C. In contrast, at 900 °C, about 90% of the injected  $C_2H_4$  decomposes, producing abundant carbon and hydrogen radicals and nearly doubling the hydrogen concentration relative to the initial feed (Fig. 6c). Similar to the conditions at 800 °C, the carbon yield improved substantially due to the enhanced precipitation rate from the abundant supply of carbon radicals. This surplus of hydrogen radicals effectively etches amorphous carbon from CNT surfaces, resulting in cleaner, higher-quality CNTs. As a result, the increased CNT crystallinity (higher  $I_G/I_D$ ) was shown at 900 °C compared to the crystallinity of CNTs produced at 800 °C (Fig. 3b).

To test the validity of the proposed CNT growth model, we used TEM to analyze the amorphous carbon deposited on CNT surfaces synthesized at  $T_p$  of 700, 800, and 900 °C, comparing their relative amounts (Fig. 7). At 700 °C, a moderate amount of amorphous carbon was observed. At 800 °C, a significant increase in amorphous carbon was detected, resulting from the higher concentration of carbon radicals. In contrast, at 900 °C, the CNT surfaces appeared visibly clean, attributed to the strong etching effect of hydrogen radicals. These findings

align well with our proposed model. We further confirmed that the D peak intensity correlates with the amount of surface-deposited amorphous carbon, supporting the corresponding changes in the  $I_G/I_D$  ratio.

## 4. Conclusions

We investigated how the thermal decomposition of  $C_2H_4$  influences crystallinity and carbon yield. Prior to evaluating the influence of thermal decomposition by adjusting  $T_p$ , we optimized the CNT growth conditions for maximizing carbon yield using BO, fixing the growth zone at 700 °C to define our standard synthesis conditions. Under these standard conditions, we then synthesized CNTs at  $T_p$  of 500, 600, 700, 800, and 900 °C, measuring both the carbon yield and the  $I_G/I_D$  ratio in each case. While these metrics remained stable at 500, 600, and 700 °C, notable variations emerged at 800 and 900 °C. To elucidate these differences, we conducted micro-GC analysis on the gases produced by the decomposition of  $C_2H_4$  in the preheating zone. Based on the gas analysis data, we proposed a mechanism illustrating how changes in gas-phase chemistry can influence both the carbon yield and the  $I_G/I_D$  ratio. This study highlights the practical importance of  $T_p$  in controlling CNT synthesis and provides valuable insight into the gas-phase dynamics underlying CNT growth.

## Conflicts of interest

There is no conflict of interest to declare in this study.

## Data availability

The authors confirm that the data supporting the findings of this study are available within the article and its ESI.† The raw



datasets generated supporting the current study are available from the corresponding author on reasonable request.

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## References

- 1 Y. Bai, R. Zhang, X. Ye, Z. Zhu, H. Xie, B. Shen, D. Cai, B. Liu, C. Zhang, Z. Jia, S. Zhang, X. Li and F. Wei, *Nanotechnol.*, 2018, **13**, 589–595.
- 2 T. W. Ebbesen, H. J. Lezec, H. Hiura, J. W. Bennett, H. F. Ghaemi and T. Thio, *Nature*, 1996, **382**, 54–56.
- 3 S. Berber, Y. K. Kwon and D. Tománek, *Phys. Rev. Lett.*, 2000, **84**, 4613–4616.
- 4 F. Wei, Q. Zhang, W. Z. Qian, H. Yu, Y. Wang, G. H. Luo, G. H. Xu and D. Z. Wang, *Powder Technol.*, 2008, **183**, 10–20.
- 5 Q. Zhang, M. Q. Zhao, J. Q. Huang, J. Q. Nie and F. Wei, *Carbon*, 2010, **48**, 1196–1209.
- 6 M. Corrias, B. Caussat, A. Ayrat, J. Durand, Y. Kihn, P. Kalck and P. Serp, *Chem. Eng. Sci.*, 2003, **58**, 4475–4482.
- 7 M. Li, S. Hachiya, Z. Chen, T. Osawa, H. Sugime and S. Noda, *Carbon*, 2021, **182**, 23–31.
- 8 W. Bai, D. Chu and Y. He, *Chem. Eng. J.*, 2021, **407**, 126792.
- 9 H. Song, D. H. Kim, C. W. Park, J. Jae, S. Hong and J. Lee, *Carbon Lett.*, 2023, **33**, 921–929.
- 10 D. H. Kim, E. Oh, S. Kim and J. Lee, *Carbon Lett.*, 2025, **35**, 199–209.
- 11 S. Shin, H. Song, Y. S. Shin, J. Lee and T. H. Seo, *Nanomaterials*, 2024, **14**, 75.
- 12 B. J. Landi, M. J. Ganter, C. D. Cress, R. A. DiLeo and R. P. Raffaele, *Energy Environ. Sci.*, 2009, **2**, 638–654.
- 13 T. Tsuji, G. Chen, M. Yamada, J. He, Y. Shimizu, H. Sakakita, K. Hata, D. N. Futaba and S. Sakurai, *Mater. Today Chem.*, 2025, **44**, 102576.
- 14 K. Enomoto, S. Kitakata, T. Yasuhara, N. Ohtake, T. Kuzumaki and Y. Mitsuda, *Appl. Phys. Lett.*, 2006, **88**, 153115.
- 15 D. E. Tsentalovich, R. J. Headrick, F. Mirri, J. Hao, N. Behabtu, C. C. Young and M. Pasquali, *ACS Appl. Mater. Interfaces*, 2017, **9**, 36189–36198.
- 16 N. Matsumoto, A. Oshima, G. Chen, M. Yudasaka, M. Yumura, K. Hata and D. N. Futaba, *Carbon*, 2015, **87**, 239–245.
- 17 P. E. Lyons, S. De, F. Blighe, V. Nicolosi, L. F. C. Pereira, M. S. Ferreira and J. N. Coleman, *J. Appl. Phys.*, 2008, **104**, 044302.
- 18 J. Lee, D. M. Lee, Y. Jung, J. Park, H. S. Lee, Y. K. Kim, C. R. Park, H. S. Jeong and S. M. Kim, *Nat. Commun.*, 2019, **10**, 1–10.
- 19 C. J. Lee, J. Park, Y. Huh and J. Yong Lee, *Chem. Phys. Lett.*, 2001, **343**, 33–38.
- 20 W. Z. Li, J. G. Wen and Z. F. Ren, *Appl. Phys. A: Mater. Sci. Process.*, 2002, **74**, 397–402.
- 21 C. J. Lee, J. Park and J. A. Yu, *Chem. Phys. Lett.*, 2002, **360**, 250–255.
- 22 H. Kimura, J. Goto, S. Yasuda, S. Sakurai, M. Yumura, D. N. Futaba and K. Hata, *Sci. Rep.*, 2013, **3**, 1–6.
- 23 R. M. Malek Abbaslou, J. Soltan and A. K. Dalai, *Appl. Catal., A*, 2010, **372**, 147–152.
- 24 S. Y. Moon, B. R. Kim, C. W. Park, S. H. Lee and S. M. Kim, *Chem. Eng. J. Adv.*, 2022, **10**, 100261.
- 25 M. Shen, T. Inoue, M. Wang, Y. Liu and Y. Kobayashi, *ACS Appl. Mater. Interfaces*, 2025, **17**, 20105–20115.
- 26 J. Lee, M. Abdulhafez and M. Bedewy, *J. Phys. Chem. C*, 2019, **123**, 28726–28738.
- 27 J. Lee, M. Abdulhafez and M. Bedewy, *Ind. Eng. Chem. Res.*, 2019, **58**, 11999–12009.
- 28 J. Lee, M. Abdulhafez and M. Bedewy, *J. Manuf. Sci. Eng.*, 2019, **141**, 1–8.
- 29 J. Lee, G. N. Tomaraei, M. Abdulhafez and M. Bedewy, *Chem. Mater.*, 2021, **33**, 6277–6289.
- 30 M. Balandat, B. Karrer, D. R. Jiang, S. Daulton, B. Letham, A. G. Wilson and E. Bakshy, *Adv. Neural Inf. Process. Syst.*, 2020, **33**, 21524–21538.
- 31 S. Burhenne, D. Jacob and G. P. Henze, *Proc. Build. Simul. 2011 12th Conf. Int. Build. Perform. Simul. Assoc.*, 2011, pp. 1816–1823.
- 32 E. R. Meshot, D. L. Plata, S. Tawfick, Y. Zhang, E. A. Verploegen and A. J. Hart, *ACS Nano*, 2009, **3**, 2477–2486.
- 33 G. D. Nessim, M. Seita, D. L. Plata, K. P. O'Brien, A. John Hart, E. R. Meshot, C. M. Reddy, P. M. Gschwend and C. V. Thompson, *Carbon*, 2011, **49**, 804–810.
- 34 H. Sugime and S. Noda, *Carbon*, 2012, **50**, 2953–2960.
- 35 F. E. Frey and D. F. Smith, *Ind. Eng. Chem.*, 1928, **20**, 948–951.
- 36 S. Hofmann, R. Blume, C. T. Wirth, M. Cantoro, R. Sharma, C. Ducati, M. Hävecker, S. Zafeirotos, P. Schnoerch, A. Oestereich, D. Teschner, M. Albrecht, A. Knop-Gericke, R. Schlögl and J. Robertson, *J. Phys. Chem. C*, 2009, **113**, 1648–1656.
- 37 S. B. Sinnott, R. Andrews, D. Qian, A. M. Rao, Z. Mao, E. C. Dickey and F. Derbyshire, *Chem. Phys. Lett.*, 1999, **315**, 25–30.

