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Photoluminescence fluctuations in single perovskite nanocrystals: structural, environmental and ligand effect

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Perovskite nanocrystals (PNCs) show great promise for optoelectronic devices; yet at the single-particle level, they are susceptible to photoluminescence (PL) fluctuations. Single-particle studies provide key insights into the photophysical processes responsible for these fluctuations. This review discusses both intrinsic factors, such as size and surface defects, and extrinsic factors, including moisture and oxygen, that contribute to PL instability in PNCs. We also highlight recent advancements in surface passivation techniques that effectively reduce or suppress the PL fluctuations, thereby enhancing the stability and optical performance of PNCs. Ultimately, understanding and mitigating PL fluctuations are essential for improving the stability and efficiency of PNC-based devices.

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Introduction

Perovskite nanocrystals (PNCs) have the chemical formula ABX_3 , where A represents an inorganic or organic cation (*i.e.*, Cs^+ , $CH_3NH_3^{3+}$ (MA^+), $CH(NH_2)_2^{2+}$ (FA^+)), B is a metal cation (*i.e.*, Pb^{2+} , Sn^{2+} , Zn^{2+}), and X is a halide anion (*i.e.*, Cl^- , Br^- , I^-). They have attracted significant attention due to their remarkable properties, including narrow emission line widths, high

photoluminescence (PL) quantum yields, and size- and composition-dependent tunable band gaps.^{1–7} These characteristics make them promising candidates for light-emitting diodes, lasers, and photovoltaic applications.^{8–16} Nonetheless, similar to traditional II–VI, III–V, and IV–VI semiconductors, at the single-particle level, PNCs exhibit PL fluctuations, where the emission intensity varies among high (ON), intermediate (gray), and low (OFF) levels over time under continuous photo-excitation. The lowering in PL intensity occurs when the photo-generated excitons (electron–hole pairs) undergo non-radiative recombination after excitation such as trapping at a defect site,

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resulting in a temporary loss of PL emission. When non-radiative processes are not activated, radiative recombination can resume, resulting in the recovery of PL intensity. Fundamentally, these PL fluctuation properties reflect the inherent structural properties of the single emitters and their microenvironment. Specifically, for semiconductor NCs, they are related to their crystal structures, defects, and surfaces.

PNCs offer greater defect tolerance compared to conventional semiconductor NCs such as CdSe and GaAs, primarily due to the higher formation energy of ion misplacement in the perovskite structure.^{4,7,17–20} As a result, the formation of interstitial and anti-site point defects is less likely, which reduces the possibility of creating deep traps (energy levels within the band gap). Nevertheless, due to their ionic nature, PNCs are not entirely free of defects and are susceptible to structural instability.^{18,21} Ion migration in the PNCs can create vacancies and cause the detachment of weakly bound surface ligands, which affects the stability of the PNCs. These vacancies then act as electron traps, increasing the rate of nonradiative recombination.^{20,22,23} Additionally, the oleylamine (OLA) and oleic acid (OA) ligands commonly used during PNC synthesis dynamically bind to and dissociate from the NC surface.^{24–26} The dynamic binding results in an unpassivated NC surface, giving rise to surface defects that act as traps and facilitate nonradiative recombination.^{22,26} Thus, although PNCs are less prone to deep trap formation due to their defect tolerance property, their ionic nature and dynamic surface chemistry contribute to PL fluctuations in these materials.

Photoluminescence fluctuation models

Different PL intensity fluctuation patterns have been observed in PNCs, known as “blinking” and “flickering”, shown in Fig. 1a and b, respectively. PL intensity “blinking” is characterized by abrupt switching between “ON” and “OFF” states, whereas “flickering” involves gradual intensity fluctuations between multiple intensity states. In addition, fluorescence lifetime intensity distribution (FLID) diagrams have been reported for PNCs. FLID diagrams are two-dimensional plots where the PL intensity and lifetime are determined and plotted, for each time bin (typically tens of milliseconds) of the PL data. These diagrams show the relationship between the PL intensity and PL lifetimes (Fig. 1c–e) and can be related to the different PL fluctuation models. Both the PL intensity time traces and FLIDs vary across the PNCs even when synthesized in the same batch. To date, several models have been proposed to explain the cause of PL fluctuations. Auger recombination has been shown to cause PL fluctuations in many kinds of semiconductor NCs (Fig. 1f).^{27–30} This phenomenon occurs due to the charging and discharging of PNCs, influenced by deep traps. These deep traps capture the electrons or holes generated after excitation, leaving behind an extra charge. When a new exciton forms, the NC enters a charged state (also known as a “trion”). In this trion state, the exciton can recombine radiatively; or more likely, the exciton undergoes a non-radiative Auger-like process, which often has a much higher rate than the rate of radiative recombination, by

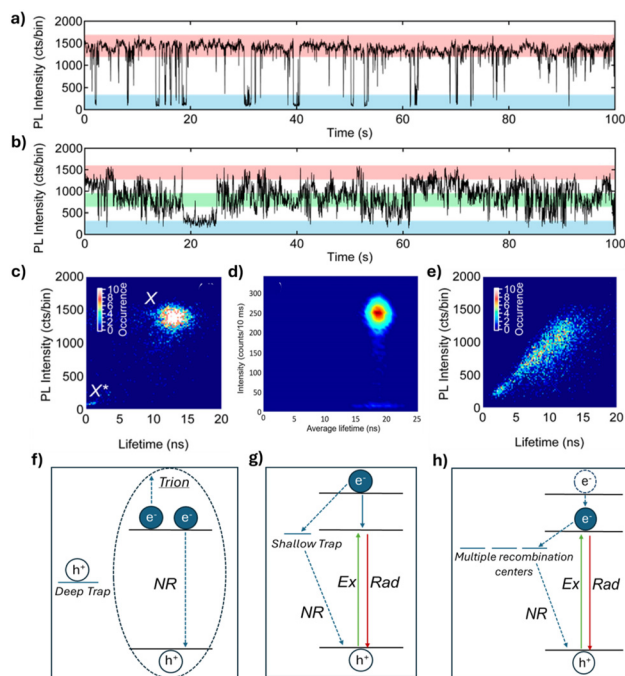


Fig. 1 Representative PL intensity traces showing (a) blinking and (b) flickering in FAPbBr₃ NCs. Reproduced from ref. 39 with permission from *American Chemical Society*, copyright 2017. (c–e) FLID diagrams of three FAPbBr₃ NCs. Reproduced from ref. 36 with permission from *American Chemical Society*, copyright 2017, and ref. 39 with permission from *Scientific Reports*, copyright 2020. (f–h) Schematic illustrations of the PL fluctuation models: (f) Auger recombination; (g) hot carrier (HC) trapping; and (h) non-radiative band-edge exciton recombination via multiple recombination centers. Where “Ex” represents excitation, “Rad” represents radiative recombination, and “NR” represents nonradiative recombination.

transferring energy to the additional charge (Fig. 1f). Compared to the high PL intensity states, the low intensity states caused by the Auger process are characterized by shorter PL lifetimes, as shown by the FLID in Fig. 1c.^{31,32} In the presence of shallow traps, hot carrier (HC) trapping and non-radiative band-edge carrier (NBC) recombination are proposed to cause PL fluctuations.³³ HC trapping occurs before the hot carrier relaxes to the band edge, where the hot charge carrier (e.g. an excited electron) rapidly relaxes through a nonradiative recombination pathway via the traps near the band-edge, before another exciton is generated (Fig. 1g). This type of non-radiative recombination leads to a decrease in PL intensity but is not accompanied by a significant change in PL lifetime.^{33–36} However, the origin of these traps remains unknown. NBC recombination, on the other hand, is proposed to occur through the activation and deactivation of multiple recombination centers. This may result from the emptying and filling of surface trap states, potentially caused by the dynamic binding of surface ligands. The shallow and short-lived traps capture charge carriers from the band-edge as they relax, resulting in nonradiative recombination (Fig. 1h). As a result, the PL intensity fluctuates gradually due to competition between a fixed



radiative rate and varying nonradiative rates in the NCs. The PL fluctuation pattern resulting from NBC recombination is characterized by gradual changes over time, known as flickering, with a gray state occurring more frequently than the ON state (Fig. 1b).^{33,34} Although FLID diagrams are presented to support the proposed models, more work is needed to clearly explain them, particularly given the complexity when analyzing flickering (as discussed later). Among the few studies that have reported flickering in PNCs, the corresponding FLID diagram typically shows a positive correlation between PL intensity and lifetime (Fig. 1e).^{37–39} While the proposed models described and shown in Fig. 1f–h represent possible recombination pathways responsible for the observed PL fluctuations in PNCs, it is important to note that a single PNC may exhibit PL fluctuations caused by a combination of these processes rather than by just one.

While PL fluctuations have been observed in PNCs, further research is needed to fully understand the mechanisms causing the fluctuations in single PNCs. PL fluctuations are more complex in PNCs, exhibiting flickering (Fig. 1b) – a behavior not commonly observed in traditional semiconductor NCs. Additionally, the ionic nature of perovskites makes them sensitive to environmental factors such as oxygen, moisture, high temperatures, and light, which can lead to irreversible phase transformations and decomposition during data acquisition, thereby limiting their experimental durations and complicating data collection and analysis.^{21,40–42} Despite these challenges, single PNC studies are sensitive to the variations between individual NCs and allow for the analysis of quenching, recombination, trapping, and the impact of external stimuli on exciton dynamics without being averaged by the ensemble. They provide a better understanding of how the various surface defects affect PNCs, which is of great importance to improve the durability and performance of PNC-based applications. This understanding can also give rise to more efficient strategies for material design, surface passivation, and device optimization. The following sections of this review will discuss recent advancements in single-particle PL studies of PNCs and examine the factors that affect their PL fluctuations (illustrated in Fig. 2). It also includes the authors' perspective on future work in this research area.

Role of polymer matrices during data collection

As previously mentioned, PNCs are highly sensitive to moisture and oxygen. Over time, this sensitivity worsens their optical properties and can lead to the ultimate decomposition of the PNCs, significantly affecting data collection. To overcome this problem, researchers have found that dispersing PNCs in a protective polymer matrix can prevent decomposition and extend the duration of data collection. Rainò *et al.* showed that the choice of polymer matrix plays a crucial role in this process.⁴³ The polymer used must be soluble in the same apolar solvents as the PNCs and have low autofluorescence. Among the four polymers tested with inorganic CsPbBr₃ NCs, *i.e.* poly(methyl methacrylate) (PMMA), cyclo-olefin copolymer (TOPAS), styrene–ethylene–butylene–styrene block copolymer

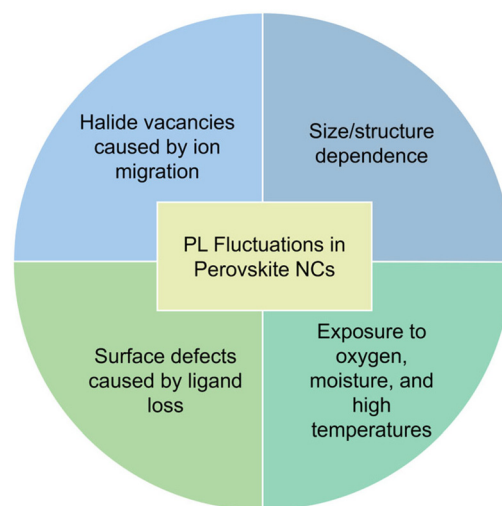


Fig. 2 Summary of factors influencing PL fluctuations in PNCs.

(SEBS), and polystyrene, polystyrene displayed the best performance by ensuring stable PL emission and suppressing PL blueshift caused by decomposition of the PNCs (Fig. 3a). This

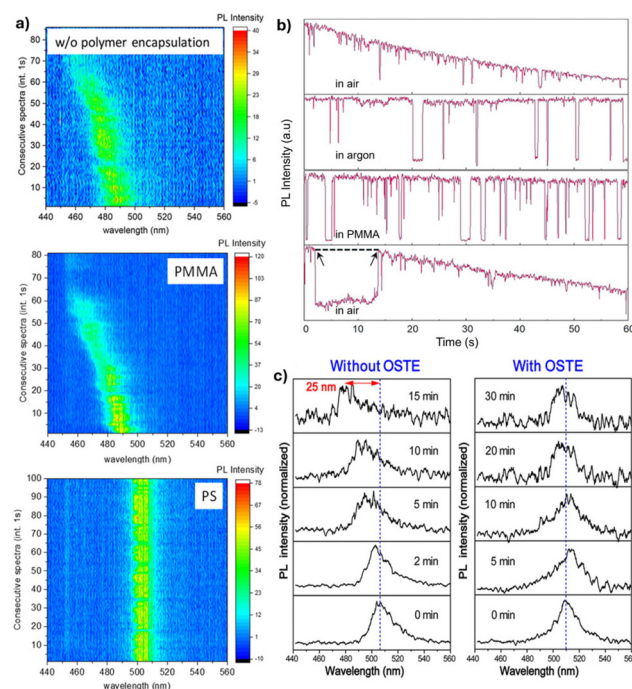


Fig. 3 (a) Two-dimensional PL spectral maps of CsPbBr₃ NCs without polymer protection and embedded in PMMA and polystyrene. Reproduced from ref. 43 with permission from *American Chemical Society*, Copyright 2019. (b) PL intensity traces of single MAPbI₃ NCs in air, argon, PMMA, and displaying a long-lived OFF state in air. Reproduced from ref. 44 with permission from *Angewandte Chemie International Edition*, Copyright 2019. (c) PL spectra evolution of single MAPbBr₃ NCs without polymer protection and with OSTE film coating, under laser exposure for 15 and 30 minutes, respectively. Reproduced from ref. 45 with permission from *American Chemical Society*, Copyright 2019.



is primarily due to its hydrophobic nature, with the aromatic rings in polystyrene showing a stronger interaction with the hydrophobic ligand shell of the CsPbBr₃ NCs.⁴³ Although the results suggested that PMMA may not be an ideal polymer matrix for PNCs, Chouhan *et al.* demonstrated that PMMA effectively maintained stable PL emission of organic–inorganic MAPbI₃ NCs (Fig. 3b).⁴⁴ Additional work has been conducted on tailoring polymer matrices to improve ligand–polymer interactions. For example, an oxygen-scavenging thiol-based polymer, off-stoichiometry thiol–ene (OSTE), was utilized to prevent photodegradation of MAPbX₃ NCs caused by oxygen.⁴⁵ As shown in Fig. 3c, the OSTE polymer encapsulates the NCs, providing a protective layer that prevents PL blueshift caused by decomposition of the NCs.^{42,45} While polymer matrices offer significant benefits in stabilizing PNCs and extending data collection time, the choice of polymer is critical and must be tailored to the specific type of NC to effectively prevent decomposition and maintain optical performance.

Influence of excitation wavelength and intensity

Extrinsic factors, such as the excitation wavelength and intensity, have been shown to influence the PL fluctuations of PNCs. In a study by Gibson *et al.*, PL intensity traces of single CsPbBr₃ NCs were collected at varying excitation intensities.⁴⁶ The traces were then analyzed by using an intensity threshold to define the “ON” (bright) and “OFF” (dark) states, and the durations (τ_i) for which the PL intensity remains in the ON and OFF states were then determined. By analyzing the probability distributions of the ON and OFF times, valuable information about the state-to-state kinetics can be obtained. Gibson *et al.* showed that in CsPbBr₃ NCs, the ON-state times can be fitted with a truncated power law, $P(\tau_i) \propto \tau_i^{-\alpha_i} e^{-\tau_i/\tau_c}$, where the truncation time τ_c , is the duration at which the power law behavior of the ON-state transitions to exponential decay.⁴⁶ The ON-state truncation time becomes shorter as the excitation intensity increases from 2.5 to 17.7 W cm⁻² until saturation (Fig. 4a). Unlike the ON-state, the OFF-state did not show a truncation time dependence on the excitation intensity. Similar results were reported for organic–inorganic FAPbBr₃ and CH₃NH₃PbBr₃ NCs.^{47,48} In addition to their dependence on excitation intensity, the durations of the ON and OFF states have also been studied at different excitation wavelengths. Since the distribution of OFF times represents the recovery from low to high intensity states, the power-law behavior in the OFF times provides insights into the trapping and de-trapping dynamics of charge carriers within NCs.^{49,50} Singha *et al.* investigated the ON and OFF distributions of FAPbBr₃ NCs excited at 400 nm and 440 nm, with power densities ranging from 1.4 to 4 W cm⁻² (Fig. 4b).³³ The results show that according to the truncated power law, an increase in excitation power density leads to a longer OFF-state duration for both 400 nm and 440 nm excitation. The key difference is that the OFF-state durations differ more significantly at 440 nm than at 400 nm, indicating a stronger dependence on the excitation power at higher wavelengths. This suggests that, at longer excitation wavelengths, the competition between HC trapping and Auger

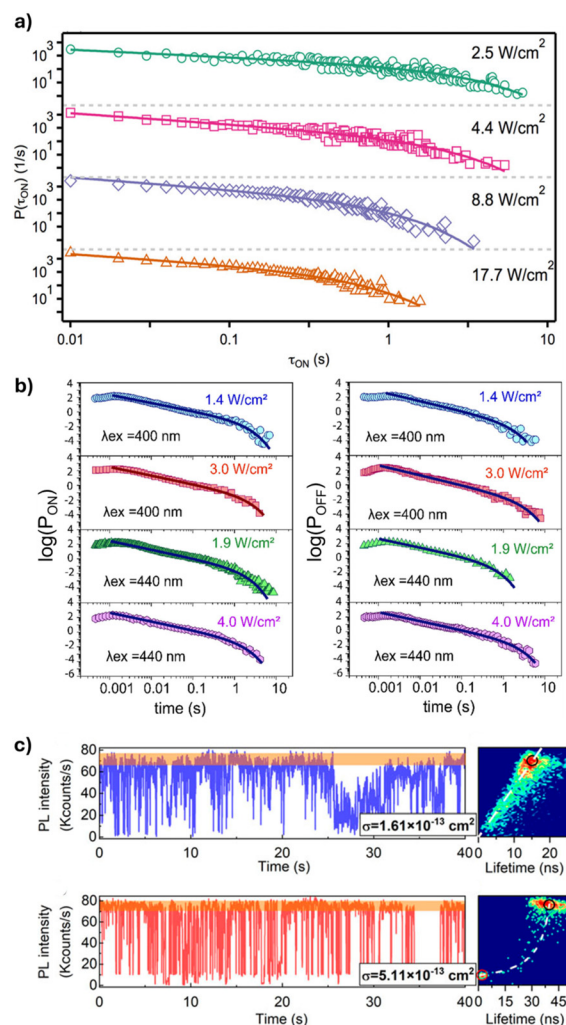


Fig. 4 (a) ON-state probability distributions of CsPbBr₃ NCs showing truncation at earlier times with increasing excitation intensity. Reproduced from ref. 46 with permission from *American Chemical Society*, Copyright 2018. (b) ON and OFF-state duration events extracted from PL intensity traces at different excitation wavelengths and intensities. Reproduced from ref. 33 with permission from the *Journal of Chemical Physics*, Copyright 2024. (c) PL fluctuation trajectories of a relatively small-sized CsPbI₃ NC ($\sigma = 1.61 \times 10^{-13}$ cm²) and a relatively large-sized CsPbI₃ NC ($\sigma = 5.11 \times 10^{-13}$ cm²), along with the corresponding FLID diagrams. Reproduced from ref. 37 with permission from the *Journal of Chemical Physics*, Copyright 2024.

processes becomes more pronounced. This is due to the fact that at longer excitation wavelengths, the excitation energy is closer to the band edge, which reduces the likelihood of carriers being trapped in deep traps. This results in faster de-trapping rates and slower trapping rates.³³ This phenomenon was also observed by Mandal *et al.*, where the ON fraction (the percentage of time the NC spends in the ON state) of CsPbBr₃ NCs increases as the excitation wavelength increases from 405 to 453 and then to 488 nm.⁵¹ This change is attributed to an increase in the ratio of carrier de-trapping rate to trapping rate at longer excitation wavelengths.



Dependence on size

The optical properties of PNCs depend not only on their composition but also on their size. In a recent study, Yang *et al.* determined the absorption cross section of single CsPbI₃ NCs by recording the PL intensity of individual NCs at varying excitation powers to generate a PL saturation curve.³⁷ The curve was then fit using the equation, $I \propto 1 - e^{-\sigma j}$, where σ is the absorption cross section and j is the excitation photon flux calculated from the laser power, repetition rate, and photon energy. They observed different PL fluctuation patterns at low excitation powers for CsPbI₃ NCs with an absorption cross section of $1.61 \times 10^{-13} \text{ cm}^{-2}$ and $5.11 \times 10^{-13} \text{ cm}^2$. The smaller NCs exhibit a series of continuously distributed emission states, and were considered to be “flickering” by the authors; while the larger NCs exhibited distinct binary ON and OFF emission states (Fig. 4c). Moreover, the PL intensity and lifetime show a linear correlation in smaller NCs and a non-linear correlation in larger NCs. Smaller NCs differ from larger ones primarily in the degree of overlap between their exciton wave functions and the electronic states associated with surface defects. In smaller NCs with sizes smaller than their exciton Bohr diameter, there is significant overlap between the two, which leads to fast trapping and de-trapping of carriers, resulting in NBC recombination *via* multiple recombination centers. In contrast, there is less overlap in larger NCs with sizes greater than the exciton Bohr diameter; therefore, trapping and de-trapping of carriers become slower and less efficient. This suggests that Auger recombination is primarily responsible for PL fluctuations in larger-sized CsPbI₃ NCs.³⁷ The influence of size in quantum-confined PNCs has also been studied. Paul *et al.* studied CsPbBr₃ NCs with three sizes, 3.80, 4.80, and 5.90 nm.⁵² These different-sized NCs exhibit similar PL intensity traces. However, the FLID diagrams reveal that in smaller CsPbBr₃ NCs, the high PL intensity states are associated with long lifetimes, while the low intensity states display either short or long lifetimes, corresponding to short- and long-lived carrier traps, respectively. In contrast, the larger size NCs showed mostly high-intensity-long-lifetime features. Smaller NCs also exhibit a lower ON-state fraction, indicating a higher carrier trapping rate and suggesting that de-trapping is more difficult in smaller NCs compared to larger ones.⁵² Similar results were reported in phenethylammonium bromide-treated CsPbBr₃ NCs with sizes ranging from 3.6 to 14 nm.⁵³ Overall, the size of PNCs plays a critical role in determining their charge carrier dynamics and PL fluctuations. Among the different-sized PNCs studied so far, smaller NCs exhibit higher trapping rates and more complex PL emission patterns, such as flickering, while larger NCs primarily display distinct ON and OFF states.

Effects of external stimuli

Due to their ionic nature, PNCs are highly sensitive to changes in the external stimuli such as oxygen, moisture, heat, and light, which can lead to irreversible phase transitions and decomposition. In particular, elevated temperatures can accel-

erate ion migration, promoting the formation of non-radiative recombination centers, while humidity can facilitate displacement of surface ligands due to phase transformations affecting the stability of PNCs.^{40,42,54,55} To investigate this effect, Hong *et al.* monitored the PL intensity of individual CsPbBr₃ NCs under various conditions, including nitrogen gas, dry air, oxygen, and moisture.⁵⁶ When the CsPbBr₃ NCs were exposed to a cycle of dry air, dry air + H₂O, and then dry air again, an increase in the PL fluctuation intensity was observed in the presence of H₂O (Fig. 5a). Specifically, the PL intensity increased at a humidity level of 40%. The increase in the PL intensity and the reduced OFF durations were attributed to the adsorption of H₂O molecules on the PNC surface, which lowers the energetic barrier of midgap halide vacancies. These vacancies act as carrier traps; by lowering their energy barrier, H₂O adsorption promotes their transition from an active state, where they capture charge carriers and cause non-radiative recombination, to a passive state, where they no longer trap charge carriers and allow radiative recombination to occur. It is worth noting that when the humidity was raised

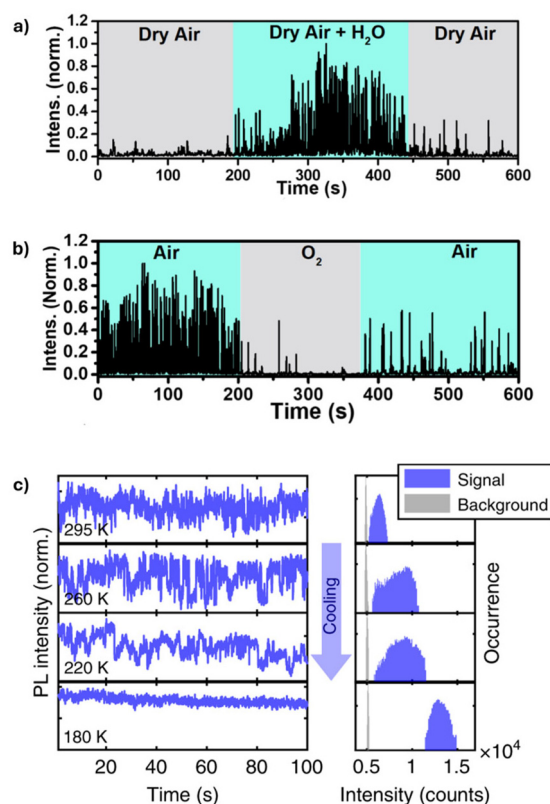


Fig. 5 (a) PL intensity trace of single CsPbBr₃ NC monitored under alternating exposure to dry air, air + H₂O, and dry air atmosphere. (b) PL intensity trace of single CsPbBr₃ NC monitored under alternating exposure to air, oxygen, and air atmosphere. Reproduced from ref. 56 with permission from *American Chemical Society*, Copyright 2022. (c) Normalized temperature-dependent PL intensity traces and corresponding intensity histograms with logarithmic vertical axis. Reproduced from ref. 59 with permission from *Nature Communications*, Copyright 2019.



to 60%, a decrease in PL intensity, followed by complete quenching, was observed, attributed to the effect of aggregated water, which is known to induce PNC decomposition. This study also showed that pure oxygen causes strong PL quenching of the PNCs (Fig. 5b). This is due to the interaction of oxygen with surface defects and traps, which accelerates the degradation of the NCs.⁵⁶ In a study by Yuan *et al.*, the effect of moisture on the PL of single CsPbI₃ PNCs in the presence of light was monitored.⁴² An increase in the PL intensity of CsPbI₃ NCs was observed when exposed to moisture, as previously observed and explained.⁵⁶ However, under continuous excitation, it was found that the CsPbI₃ NCs lost PL emission within 5–10 min, indicating that light accelerates the degradation process. Additionally, a blue shift was observed in the PL spectra of the CsPbI₃ NCs when exposed to either moisture or continuous laser excitation, indicating a decrease in the size of the NCs due to decomposition.⁴² While the aforementioned studies focus on quantum-confined PNCs, similar degradation behavior in the presence of moisture and oxygen has been observed in MAPbI₃ single crystals, approximately 800 nm in size, which also exhibit PL fluctuations.⁵⁷

Temperature-dependent PL properties have been observed in CdSe-based NCs and also in PNCs.^{6,31,58} The PL quantum yield of organo-metal PNCs has been shown to increase at lower temperatures, indicating reduced nonradiative recombination. Gerhard *et al.* studied the temperature-dependent PL properties of single MAPbI₃ NCs and observed a progressive reduction in PL intensity fluctuation upon cooling from 300 K to 77 K (Fig. 5c), with the most pronounced reduction occurring below 200 K.⁵⁹ This reduction in PL fluctuation is attributed to the nonradiative channels switching from an active to a passive state at lower temperatures. This switching is driven by thermal barriers in the range of 0.2–0.8 eV. At lower temperatures, ion migration is less likely to occur due to the high activation barrier, which leads to reduced PL fluctuation and increased PL intensity. At higher temperatures, ions can easily overcome the activation barrier, leading to ion migration. This migration generates defects such as vacancies or interstitials, which act as traps, resulting in increased non-radiative recombination.⁵⁹ Similarly, Rainò *et al.* showed that at low temperatures of 6 K, CsPb(Cl/Br)₃ shows good photostability without surface passivation.⁶⁰ In general, the sensitivity of PNCs to environmental factors such as humidity, oxygen, and temperature significantly impacts their PL behavior and stability. Understanding these effects, such as moisture-induced PL enhancement/quenching and oxygen-induced quenching, provides valuable insights into the degradation mechanisms of PNCs and highlights the importance of environmental control for optimizing their optical performance.

Achieving PL fluctuation suppression through surface passivation

Although the atomistic nature of the surface states in PNCs is often unknown, several studies have shown that surface treatment can significantly improve the optical properties of PNCs, primarily by reducing surface defects caused by vacancies and

desorption of surface ligands, which create vulnerable sites for moisture and oxygen to react with the PNCs.⁴ The strategies include encapsulation with an inorganic shell,^{61,62} surface treatment through the addition of excess halide salts or pseudohalogens to passivate any vacancies,^{23,34,63–65} and using alternative ligands that have a stronger binding affinity to the NC surface.^{38,66,67}

One effective method of surface passivation is through shell growth. The shell can serve as a protective barrier against environmental factors, such as oxygen and moisture, and improve the optical properties of NCs.^{5,17} Guo *et al.* demonstrate improved PL properties of Cs₄PbBr₆ NCs through the encapsulation using alumina. The alumina-coated Cs₄PbBr₆ NCs show reduced PL intensity fluctuation and improved stability due to the passivation of surface defects. Additionally, delayed emission was observed, resulting from charge trapping, storage, and subsequent recovery to the emissive manifold.⁶¹ Tang *et al.* encapsulated CsPbBr₃ NCs with CdS to reduce the occurrence of nonradiative Auger recombination.⁶² The CsPbBr₃/CdS core/shell NCs exhibited an average ON fraction larger than 99% with little to no grey states, indicating reduced deep-trap formation due to the CdS shell.

In addition to shell growth, passivation of surface states can also be achieved through filling the halide vacancies caused by ion migration or the loss of capping ligands.^{22,68} The halide vacancies formed on the NC surface act as electron traps, increasing the non-radiative rate. Surface passivation can be achieved through the addition of halide salts. Park *et al.* investigated the PL fluctuation properties of CsPbBr₃ NCs modified with ZnBr₂. The modified NCs exhibited a low charge trapping rate compared to pristine CsPbBr₃ NCs. The excess bromide increased the surface Br[−] ratio and resulted in improved surface capping and suppressed PL fluctuation.⁶³ Similar observations have been made by Chouhan *et al.* by adding MABr and MAI during data collection to suppress PL intensity fluctuations of single MAPbX₃ NCs (Fig. 6a).²³ In addition to halide salts, pseudohalogens, which resemble the chemistry of true halogens, can be used to passivate halide vacancies.^{34,39} Yarita *et al.* observed the impact of sodium thiocyanate (NaSCN) on the PL fluctuation of single FaPbBr₃ NCs. Of the two types of PL fluctuation patterns observed, blinking and flickering, flickering behavior was completely suppressed while blinking remained unchanged (Fig. 6b), suggesting that surface states or the surrounding environment play an important role in the origin of flickering as opposed to charging/discharging of the NCs.³⁹

Due to the dynamic binding of oleylamine (OLA) and oleic acid (OA) ligands that are commonly used in the synthesis of PNCs, the resulting NC surface is prone to disorder and defect formation. Proton transfer is required to transform OA and OLA into their ionic forms, which act as capping ligands for PNCs. While the oleate anion binds strongly to surface lead cations, the oleylammonium cation interacts weakly with surface halides, resulting in a dynamic equilibrium that promotes ligand desorption and leads to an unpassivated surface.^{69,70} To mitigate this issue, alternative ligands with a



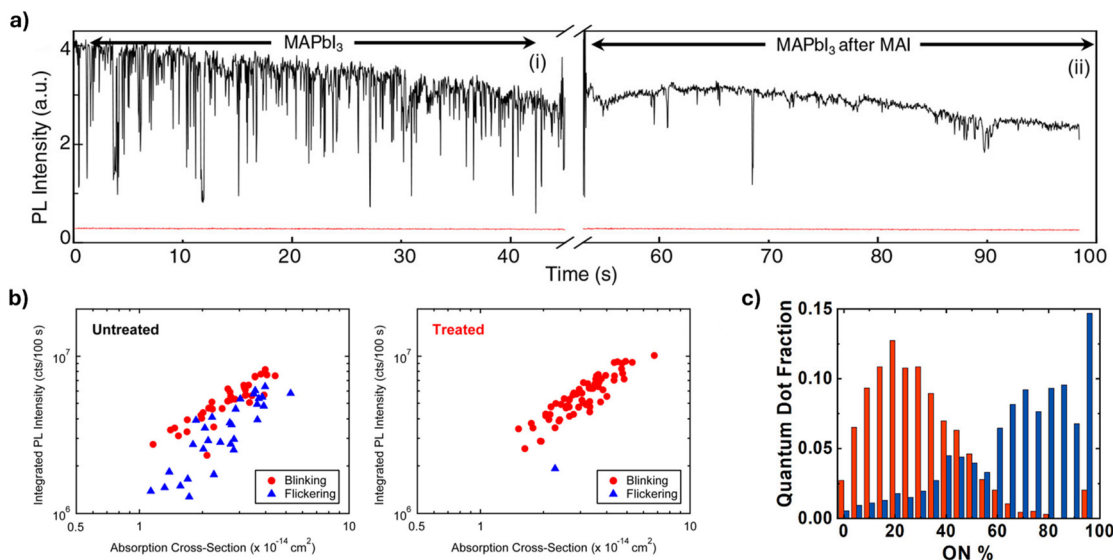


Fig. 6 (a) PL intensity trajectories showing fluctuation suppression of MAPbI₃ NCs (i) before and (ii) after treatment with a MAI solution. Reproduced from ref. 23 with permission from *American Chemical Society*, Copyright 2021. (b) Time-integrated PL intensity for the low PL intensity region plotted for untreated and sodium thiocyanate (NaSCN) treated single FAPbBr₃ NCs. Reproduced from ref. 39 with permission from *American Chemical Society*, Copyright 2017. (c) Distribution of the ON percentages for oleic acid/oleylamine (red) and lecithin (blue) capped CsPbBr₃ NCs. Reproduced from ref. 66 with permission from *American Chemical Society*, Copyright 2024.

stronger binding affinity to the NC surface have been employed. Recent studies have explored amine-free synthesis routes, Praneeth *et al.* demonstrated that replacing oleylamine with trioctylphosphine as a capping ligand in the synthesis of CsPbBr₃ NCs resulted in amine-free PNCs exhibiting stable, non-blinking PL with no long-lived OFF states.⁷¹ This is attributed to reduced nonradiative recombination due to more efficient surface passivation by trioctylphosphine compared to oleylamine. Gallagher *et al.* investigated how changes in ligand equilibrium affect the PL properties of lecithin-capped CsPbBr₃ NCs compared to those capped with OA/OLA ligands.⁶⁶ Lecithin-capped NCs exhibited more stable PL emission, spending on average 68% of the time in the ON state compared to 30% for OA/OLA capped NCs, and showed a higher probability of staying in the ON state (Fig. 6c). This enhanced stability is attributed to the stronger binding affinity of lecithin to the CsPbBr₃ NC surface, which better preserves surface integrity. In contrast, OA/OLA ligands are more weakly bound and tend to detach during the dilution process used to prepare single-particle samples, leading to surface degradation and less stable emission.⁶⁶ Comparable results were reported by Kuang *et al.*, who found that the density of trap states increases as the surface ligands of CsPbBr₃ NCs decrease.⁷² Similarly, alkythiols such as ethanethiol have also been utilized as an alternative capping ligand for CsPbBr₃ NCs. Seth *et al.* observed both blinking and flickering PL fluctuation behaviors in single CsPbBr₃ NCs. Treating these NCs with ethanethiol converts the flickering of some NCs to blinking, but no change was observed for the blinking ones. The reduction of flickering is attributed to the passivation of uncoordinated lead atoms on the NC surface, which serve as

shallow electron traps.³⁸ These findings highlight the significant impact of surface treatment and passivation strategies on the optical properties and stability of PNCs. By employing methods such as shell growth, halide salt addition, and alternative ligands, suppressed PL fluctuations and enhanced stability can be achieved.

Conclusions and future perspectives

In summary, single-particle studies have greatly improved our understanding of the stability and degradation mechanisms in PNCs, particularly with regard to photoinduced degradation and ion migration. By monitoring the optical behavior of individual NCs, these studies have provided valuable insights into how external factors like light exposure, temperature, and moisture contribute to the degradation process. Photoinduced degradation, often marked by PL quenching and spectral shifts, has been shown to reduce the photostability of PNCs over time. Ion migration, driven by factors such as temperature and humidity, exacerbates this issue by creating surface defects, which in turn lead to non-radiative recombination and PL fluctuations. These studies also highlight potential solutions, such as surface passivation through encapsulation or ligand modification, which can mitigate the impact of these degradation processes and improve the stability of PNCs. Single PNC studies deepen our understanding of fundamental degradation pathways, thereby enabling the development of more stable and efficient PNC-based materials for a wide range of optoelectronic applications.



Experimental challenges arise when considering the impact of environmental factors, such as moisture and oxygen, on the measurements of single PNCs. Although various strategies have been developed to improve the stability of PNCs for single particle studies, there is still a clear demand for methods to achieve long-term stability of PNCs in air or aqueous environments under photoexcitation, enabling their applications in single particle tracking in biological systems. Additionally, low laser power is typically required for long acquisition times in single PNC studies to reduce photodegradation. However, this results in a lower PL signal, reducing the signal-to-noise ratio and photon counts, which can make it difficult challenging to distinguish between different intensity states, a difficulty further compounded by “flickering”. “Flickering” is characterized by continuous variation of PL intensity over time but has not been clearly defined to date. This qualitative description of PL intensity “flickering” makes it harder to identify distinct intensity states and to understand the mechanism causing this behavior.⁴⁶ To address this, statistical methods such as change point analysis have been attempted.^{33,46,66,73} However, caution is needed when applying these methods, as the quality of the data can heavily influence the results. Customized statistical analysis tools have also been developed in analyzing single PNCs. For example, Gallagher *et al.* integrated unsupervised clustering with change point analysis to classify CsPbBr₃ NCs based on the number of discrete intensity levels in their PL intensity traces.⁶⁶ This approach also enabled the classification of ON and OFF states. Simulated PL traces were used for validating the custom change point analysis package, which demonstrated that the method can reliably resolve up to five distinct intensity states. This shows that the method could be confidently used to analyze experimental PL fluctuation data of single PNCs. Beyond statistical analysis tools, machine learning tools have also been developed to track fluorescence trajectories in single molecules.⁷⁴ The algorithm automatically segments and clusters data without prior assumptions, identifying patterns in complex datasets. Such tools can be utilized in the analysis of PNCs given their complex nature. While statistical methods and machine learning offer promising tools for resolving intensity states of single PNCs, their effectiveness is limited by data quality, amount of data, and low PL signal level, which can lead to underestimation of the number of intensity states or intensity state changes being missed or wrongly identified.⁷³ Purely data-driven machine learning or statistical approaches may lack the interpretability of physical processes in PNCs, while physics-informed models offer mechanistic insights into the exciton lifecycles in PNCs. Thus, integrating these approaches with physics-informed models will be essential to connect single-particle data with fundamental charge carrier dynamics in PNCs.⁷⁵

In addition to the challenge of analyzing PL intensity fluctuations, the FLID diagrams observed in single PNCs vary from NC-to-NC, even if they exhibit similar PL intensity fluctuations. Fig. 7a and b are examples of FLIDs for “blinking” and “flickering” PNCs and they differ significantly from the ones in Fig. 1c–e.³⁸ Thus, a deeper understanding of the origins of

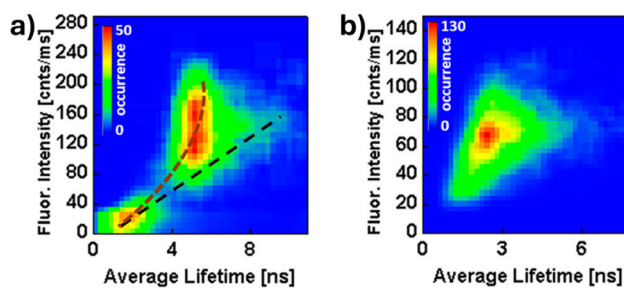


Fig. 7 FLID diagrams with false color representation of single CsPbBr₃ NCs exhibiting (a) blinking and (b) flickering. Reproduced from ref. 38 with permission from American Chemical Society, Copyright 2018.

shallow *versus* deep traps, particularly in relation to surface chemistry and defect types, is crucial for advancing mechanistic insights. Shallow traps are often associated with processes such as dynamic ligand binding or the presence of uncoordinated lead atoms, are believed to cause “flickering”. In contrast, deep traps, which may arise from intrinsic lattice defects such as vacancies or interstitials, are more likely to cause distinct OFF states, characteristic of “blinking”. Surface chemistry, including the presence of moisture, oxygen, and dynamic ligand passivation, can influence the formation of both shallow and deep traps, which can affect the observed PL fluctuation patterns. For example, a halide vacancy that would act as a deep trap in an unpassivated PNC may not lead to trap formation in the presence of strongly binding ligands such as sulfonic or phosphonic acids.^{76,77} A hybrid approach that considers both the influence of surface-induced shallow traps and intrinsic deep traps may provide a more comprehensive understanding of PL fluctuations in PNCs. To recognize the connections between PL intensity trajectories and FLIDs, advanced statistical methods and computational approaches could be considered and developed.

Ever since single-particle PL fluctuations were observed in single semiconductor NCs, the mechanistic understanding of these phenomena has always been of great interest. Although several models have been proposed to explain PL intermittency in PNCs (Fig. 1f–h), the nature of the traps (such as deep *vs.* shallow) is still unknown, especially at the atomistic scale. Using correlated multi-modal measurements (*e.g.*, combining PL with electron microscopy), it is possible to link PL emission behavior to structural dynamics at the single-particle level. However, such measurements require complicated and expensive instrumentation and special technical skills. Moreover, the changes in the structure of PNCs upon exposure to electron beams cannot be ignored. Alternatively, atomic-level computations may provide useful information about the defects in PNCs and how they influence the electronic band structures. It remains a challenge to correlate these properties with exciton dynamics in single PNCs. While single-particle studies provide valuable insights that are not detectable at the ensemble level, combining them with ensemble measurements offers additional benefits.^{78,79} For instance, Yarita *et al.* employed



femtosecond transient-absorption spectroscopy, time-resolved PL spectroscopy to study PNCs at the ensemble level, and second-order photon correlation spectroscopy on individual CsPbBr₃ NCs.⁷⁹ This approach allowed them to gain a deeper understanding of the behaviors of excitons, charged excitons, and biexcitons. Future studies may integrate single-particle and ensemble measurements to gain a more comprehensive understanding of the optical properties of PNCs. Moving beyond isolated conditions, it is highly valuable to probe PNCs in device-like conditions or working devices (e.g., LEDs or solar cells) to obtain the exciton dynamics of PNCs *in operando*.

Author contributions

H. N. prepared the manuscript under the guidance of J. Z. Both authors have approved the final version of the manuscript.

Data availability

No primary research results, software or code have been included and no new data were generated or analyzed as part of this review.

Conflicts of interest

The authors declare no competing financial interest.

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References

- 1 L. Protesescu, S. Yakunin, M. I. Bodnarchuk, F. Krieg, R. Caputo, C. H. Hendon, R. X. Yang, A. Walsh and M. V. Kovalenko, *Nano Lett.*, 2015, **15**, 3692–3696.
- 2 H. Zhu, T. Šverko, J. Zhang, D. B. Berksky, W. Sun, C. J. Krajewska and M. G. Bawendi, *Nano Lett.*, 2022, **22**, 8355–8362.
- 3 M. C. Brennan, A. Forde, M. Zhukovskiy, A. J. Baublis, Y. V. Morozov, S. Zhang, Z. Zhang, D. S. Kilin and M. Kuno, *J. Phys. Chem. Lett.*, 2020, **11**, 4937–4944.
- 4 S. Seth, T. Ahmed, A. De and A. Samanta, *ACS Energy Lett.*, 2019, **4**, 1610–1618.
- 5 T. K. T. Tran, H. N. Nyiera, C. Brea, S. N. Ruiz, C. Wang, H. Tan, S. L. Suib, G. Hu and J. Zhao, *ACS Appl. Nano Mater.*, 2024, **7**, 12153–12162.
- 6 H. Utzat, W. Sun, A. E. K. Kaplan, F. Krieg, M. Ginterseder, B. Spokoiny, N. D. Klein, K. E. Shulenberger, C. F. Perkinson, M. V. Kovalenko and M. G. Bawendi, *Science*, 2019, **363**, 1068–1072.
- 7 H. Huang, M. I. Bodnarchuk, S. V. Kershaw, M. V. Kovalenko and A. L. Rogach, *ACS Energy Lett.*, 2017, **2**, 2071–2083.
- 8 C. Y. Huang, C. Zou, C. Mao, K. L. Corp, Y. C. Yao, Y. J. Lee, C. W. Schlenker, A. K. Y. Jen and L. Y. Lin, *ACS Photonics*, 2017, **4**, 2281–2289.
- 9 Q. Zhang, R. Su, W. Du, X. Liu, L. Zhao, S. T. Ha and Q. Xiong, *Small Methods*, 2017, **1**, 1700163.
- 10 J. Tian, Q. Y. Tan, Y. Wang, Y. Yang, G. Yuan, G. Adamo and C. Soci, *Nat. Commun.*, 2023, **14**, 1433.
- 11 M. Hao, S. Ding, S. Gaznaghi, H. Cheng and L. Wang, *ACS Energy Lett.*, 2024, **9**, 308–322.
- 12 C. Wang, W. Meng, Y. Li, G. Xu, M. Peng, S. Nie and Z. Deng, *Nanoscale*, 2023, **15**, 1661.
- 13 Y. Dong, Y.-K. Wang, F. Yuan, A. Johnston, Y. Liu, D. Ma, M.-J. Choi, B. Chen, M. Chekini, S.-W. Baek, L. K. Sagar, J. Fan, Y. Hou, M. Wu, S. Lee, B. Sun, S. Hoogland, R. Quintero-Bermudez, H. Ebe, P. Todorovic, F. Dinic, P. Li, H. K. Kung, M. I. Saidaminov, E. Kumacheva, E. Spiecker, L.-S. Liao, O. Voznyy, Z.-H. Lu and E. H. Sargent, *Nat. Nanotechnol.*, 2020, **15**, 668–674.
- 14 X. Li, K. Zhang, J. Li, J. Chen, Y. Wu, K. Liu, J. Song, H. Zeng, X. Li, K. Zhang, J. Li, J. Chen, Y. Wu, K. Liu, J. Song and H. Zeng, *Adv. Mater. Interfaces*, 2018, **5**, 1800010.
- 15 H. S. Jung and N.-G. Park, *Small*, 2015, **11**, 10–25.
- 16 J. Song, J. Li, X. Li, L. Xu, Y. Dong and H. Zeng, *Adv. Mater.*, 2015, **27**, 7162–7167.
- 17 T. K. T. Tran, H. N. Nyiera and J. Zhao, *Nano Res.*, 2024, **17**, 10607–10619.
- 18 I. du Fossé, J. T. Mulder, G. Almeida, A. G. M. Spruit, I. Infante, F. C. Grozema and A. J. Houtepen, *J. Am. Chem. Soc.*, 2022, **144**, 11059–11063.
- 19 D. Meggiolaro, S. G. Motti, E. Mosconi, A. J. Barker, J. Ball, C. Andrea, R. Perini, F. Deschler, A. Petrozza and F. De Angelis, *Energy Environ. Sci.*, 2018, **11**, 702–713.
- 20 H. Jin, E. Debroye, M. Keshavarz, I. G. Scheblykin, M. B. J. Roefsaers, J. Hofkens and J. A. Steele, *Mater. Horiz.*, 2020, **7**, 397–410.
- 21 B. P. Kore, M. Jamshidi and J. M. Gardner, *Mater. Adv.*, 2024, **5**, 2200–2217.
- 22 D. Kim, T. Yun, S. An and C. L. Lee, *Nano Converg.*, 2024, **11**, 4.
- 23 V. Biju, L. Chouhan, S. Ito, E. M. Thomas, Y. Takano, S. Ghimire and H. Miyasaka, *ACS Nano*, 2021, **15**, 2831–2838.
- 24 Q. A. Akkerman, G. Rainò, M. V. Kovalenko and L. Manna, *Nat. Mater.*, 2018, **17**, 394–405.
- 25 N. Fiuza-Maneiro, K. Sun, I. López-Fernández, S. Gómez-Graña, P. Müller-Buschbaum and L. Polavarapu, *ACS Energy Lett.*, 2023, **8**, 1152–1191.
- 26 J. De Roo, M. Iba, P. Geiregat, G. Nedelcu, W. Walravens, J. Maes, J. C. Martins, I. Van Driessche, M. V. Kovalenko and Z. Hens, *ACS Nano*, 2016, **10**, 2071–2081.



- 27 R. Vaxenburg, A. Rodina, A. Shabaev, E. Lifshitz and A. L. Efros, *Nano Lett.*, 2015, **15**, 2092–2098.
- 28 A. T. Nguyen, P. Cavanaugh, I. J. La Plante, C. Ippen, R. Ma and D. F. Kelley, *J. Phys. Chem. C*, 2021, **125**, 15405–15414.
- 29 V. I. Klimov, *J. Phys. Chem. B*, 2006, **110**, 16827–16845.
- 30 M. Nirmal, B. O. Dabbousi, M. G. Bawendi, J. J. Macklin, J. K. Trautman, T. D. Harris and L. E. Brus, *Nature*, 1996, **383**, 802–804.
- 31 A. L. Efros and D. J. Nesbitt, *Nat. Nanotechnol.*, 2016, **11**, 661–671.
- 32 C. Galland, Y. Ghosh, A. Steinbrück, M. Sykora, J. A. Hollingsworth, V. I. Klimov and H. Htoon, *Nature*, 2011, **479**, 203–207.
- 33 P. K. Singha, T. Mukhopadhyay, E. Tarif, F. Ali and A. Datta, *J. Chem. Phys.*, 2024, **161**, 054704.
- 34 T. Ahmed, S. Seth and A. Samanta, *ACS Nano*, 2019, **13**, 13537–13544.
- 35 G. Yuan, D. E. Gómez, N. Kirkwood, K. Boldt and P. Mulvaney, *ACS Nano*, 2018, **12**, 3397–3405.
- 36 C. T. Trinh, D. N. Minh, K. J. Ahn, Y. Kang and K. G. Lee, *Sci. Rep.*, 2020, **10**, 2172.
- 37 C. Yang, G. Zhang, Y. Gao, B. Li, X. Han, J. Li, M. Zhang, Z. Chen, Y. Wei, R. Chen, C. Qin, J. Hu, Z. Yang, G. Zeng, L. Xiao and S. Jia, *J. Chem. Phys.*, 2024, **160**, 174505.
- 38 S. Seth, T. Ahmed and A. Samanta, *J. Phys. Chem. Lett.*, 2018, **9**, 7007–7014.
- 39 N. Yarita, H. Tahara, M. Saruyama, T. Kawawaki, R. Sato, T. Teranishi and Y. Kanemitsu, *J. Phys. Chem. Lett.*, 2017, **8**, 6041–6047.
- 40 S. Cheng and H. Zhong, *J. Phys. Chem. Lett.*, 2022, **13**, 2281–2290.
- 41 T. D. Siegler, W. A. Dunlap-Shohl, Y. Meng, Y. Yang, W. F. Kau, P. P. Sunkari, C. E. Tsai, Z. J. Armstrong, Y. C. Chen, D. A. C. Beck, M. Meilă and H. W. Hillhouse, *J. Am. Chem. Soc.*, 2022, **144**, 5552–5561.
- 42 G. Yuan, C. Ritchie, M. Ritter, S. Murphy, D. E. Gómez and P. Mulvaney, *J. Phys. Chem. C*, 2018, **122**, 13407–13415.
- 43 G. Rainò, A. Landuyt, F. Krieg, C. Bernasconi, S. T. Ochsenein, D. N. Dirin, M. I. Bodnarchuk and M. V. Kovalenko, *Nano Lett.*, 2019, **19**, 3648–3653.
- 44 L. Chouhan, S. Ghimire and V. Biju, *Angew. Chem., Int. Ed.*, 2019, **58**, 4875–4879.
- 45 L. Liu, L. Deng, S. Huang, P. Zhang, J. Linnros, H. Zhong and I. Sychugov, *J. Phys. Chem. Lett.*, 2019, **10**, 864–869.
- 46 N. A. Gibson, B. A. Koscher, A. P. Alivisatos and S. R. Leone, *J. Phys. Chem. C*, 2018, **122**, 12106–12113.
- 47 C. T. Trinh, D. N. Minh, K. J. Ahn, Y. Kang and K.-G. Lee, *ACS Photonics*, 2018, **5**, 4937–4943.
- 48 X. Han, G. Zhang, B. Li, C. Yang, W. Guo, X. Bai, P. Huang, R. Chen, C. Qin, J. Hu, Y. Ma, H. Zhong, L. Xiao and S. Jia, *Small*, 2020, **16**, 2005435.
- 49 M. Kuno, D. P. Fromm, H. F. Hamann, A. Gallagher and D. J. Nesbitt, *J. Chem. Phys.*, 2000, **112**, 3117–3120.
- 50 M. Kuno, D. P. Fromm, H. F. Hamann, A. Gallagher and D. J. Nesbitt, *J. Chem. Phys.*, 2001, **115**, 1028–1040.
- 51 S. Mandal, S. Mukherjee, C. K. De, D. Roy, S. Ghosh and P. K. Mandal, *J. Phys. Chem. Lett.*, 2020, **11**, 1702–1707.
- 52 S. Paul, G. Kishore and A. Samanta, *J. Phys. Chem. C*, 2023, **127**, 10207–10214.
- 53 C. Mi, G. C. Gee, C. W. Lander, D. Shin, M. L. Atteberry, N. G. Akhmedov, L. Hidayatova, J. D. DiCenso, W. T. Yip, B. Chen, Y. Shao and Y. Dong, *Nat. Commun.*, 2025, **16**, 204.
- 54 X. Yuan, P. Jing, J. Li, M. Wei, J. Hua, J. Zhao, L. Tian and J. Li, *RSC Adv.*, 2016, **6**, 78311–78316.
- 55 M. H. Miah, M. B. Rahman, M. Nur-E-Alam, M. A. Islam, M. Shahinuzzaman, M. R. Rahman, M. H. Ullah and M. U. Khandaker, *RSC Adv.*, 2025, **15**, 628–654.
- 56 D. Hong, Y. Zhang, S. Pan, H. Liu, W. Mao, Z. Lu and Y. Tian, *J. Phys. Chem. Lett.*, 2022, **13**, 10751–10758.
- 57 A. Halder, N. Pathoor, A. Chowdhury and S. K. Sarkar, *J. Phys. Chem. C*, 2018, **122**, 15133–15139.
- 58 S. A. Empedocles, D. J. Norris and M. G. Bawendi, *Phys. Rev. Lett.*, 1996, **77**, 3873–3876.
- 59 M. Gerhard, B. Louis, R. Camacho, A. Merdasa, J. Li, A. Kiligaridis, A. Dobrovolsky, J. Hofkens and I. G. Scheblykin, *Nat. Commun.*, 2019, **10**, 1698.
- 60 G. Rainò, G. Nedelcu, L. Protesescu, M. I. Bodnarchuk, M. V. Kovalenko, R. F. Mahrt and T. Stöferle, *ACS Nano*, 2016, **10**, 2485–2490.
- 61 T. Guo, R. Bose, X. Zhou, Y. N. Gartstein, H. Yang, S. Kwon, M. J. Kim, M. Lutfullin, L. Sinatra, I. Gereige, A. Al-Saggaf, O. M. Bakr, O. F. Mohammed and A. V. Malko, *J. Phys. Chem. Lett.*, 2019, **10**, 6780–6787.
- 62 X. Tang, J. Yang, S. Li, Z. Liu, Z. Hu, J. Hao, J. Du, Y. Leng, H. Qin, X. Lin, Y. Lin, Y. Tian, M. Zhou and Q. Xiong, *Adv. Sci.*, 2019, **6**, 1900412.
- 63 J. Park, Y. Kim, S. Ham, J. Y. Woo, T. Kim, S. Jeong and D. Kim, *Nanoscale*, 2020, **12**, 1563–1570.
- 64 H. Jin, J. A. Steele, R. Cheng, N. Parveen, M. B. J. Roeffaers, J. Hofkens and E. Debroye, *Adv. Opt. Mater.*, 2021, **9**, 2002240.
- 65 S. Paul and A. Samanta, *J. Phys. Chem. Lett.*, 2022, **13**, 5742–5750.
- 66 S. Gallagher, J. Kline, F. Jahanbakhshi, J. C. Sadighian, I. Lyons, G. Shen, B. F. Hammel, S. Yazdi, G. Dukovic, A. M. Rappe and D. S. Ginger, *ACS Nano*, 2024, **18**, 19208–19219.
- 67 C. Mi, M. L. Atteberry, V. Mapara, L. Hidayatova, G. C. Gee, M. Furis, W. T. Yip, B. Weng and Y. Dong, *J. Phys. Chem. Lett.*, 2023, **14**, 5466–5474.
- 68 Y. Wu, C. Wei, X. Li, Y. Li, S. Qiu, W. Shen, B. Cai, Z. Sun, D. Yang, Z. Deng and H. Zeng, *ACS Energy Lett.*, 2018, **3**, 2030–2037.
- 69 S. Akhil, S. Biswas, M. Palabathuni, R. Singh and N. Mishra, *J. Phys. Chem. Lett.*, 2022, **13**, 9480–9493.
- 70 F. Krieg, S. T. Ochsenein, S. Yakunin, S. Ten Brinck, P. Aellen, A. Süess, B. Clerc, D. Guggisberg, O. Nazarenko, Y. Shynkarenko, S. Kumar, C. J. Shih, I. Infante and M. V. Kovalenko, *ACS Energy Lett.*, 2018, **3**, 641–646.
- 71 N. V. S. Praneeth, S. Akhil, A. Mukherjee, S. Seth, S. Khatua and N. Mishra, *Adv. Opt. Mater.*, 2024, **12**, 2303222.
- 72 Y. Kuang, C. Zhu, W. He, X. Wang, Y. He, X. Ran and L. Guo, *J. Phys. Chem. C*, 2020, **124**, 23905–23912.



- 73 I. M. Palstra and A. F. Koenderink, *J. Phys. Chem. C*, 2021, **125**, 12050–12060.
- 74 D. S. White, M. P. Goldschen-Ohm, R. H. Goldsmith and B. Chanda, *eLife*, 2020, **9**, 53357.
- 75 Y. Sun and J. Zhao, *J. Phys. Chem. C*, 2021, **125**, 1171–1179.
- 76 N. Fiuza-Maneiro, J. Ye, S. K. Sharma, S. Chakraborty, S. Gómez-Graña, R. L. Z. Hoye and L. Polavarapu, *ACS Energy Lett.*, 2025, **10**, 1623–1632.
- 77 J. Ran, B. Wang, Y. Wu, D. Liu, C. M. Perez, A. S. Vasenko and O. V. Prezhdo, *J. Phys. Chem. Lett.*, 2023, **14**, 6028–6036.
- 78 T. Kim, S. Il Jung, S. Ham, H. Chung and D. Kim, *Small*, 2019, **15**, 1900355.
- 79 N. Yarita, H. Tahara, T. Ihara, T. Kawawaki, R. Sato, M. Saruyama, T. Teranishi and Y. Kanemitsu, *J. Phys. Chem. Lett.*, 2017, **8**, 1413–1418.

