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Exfoliated MoS₂ nanosheets immobilized in porous microbeads as recoverable photocatalysts†

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Molybdenum disulfide (MoS₂) is a highly effective visible light photocatalyst when used as well-exfoliated 2D nanosheets. The ability to make effective use of these properties is significantly compromised by the challenge of preventing nanosheet aggregation or restacking in fluid suspensions. We report a strategy for immobilizing chemically exfoliated MoS₂ as single- and few-layer nanosheets in porous crosslinked polymers prepared as microbeads. The polymeric support prevents aggregation of the nanosheets while allowing access to the nanomaterial for model organic compounds present in the surrounding fluid. Exposure to visible light results in high degradation yields (>99%) of these organic species in aqueous media, and the MoS₂ nanosheets maintain their photocatalytic efficacy through multiple cycles of use. The recoverability of the porous beads and the persistent photocatalytic activity of the polymer-supported MoS₂ offer the potential of realizing an effective, environmentally sustainable platform for photocatalytic degradation of dissolved solutes.

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Introduction

Semiconducting nanomaterials have gained widespread utilization for the photocatalytic degradation of contaminants in aqueous fluids, including dyes,^{1–3} antibiotics,^{4,5} and bacteria.^{6,7} Typically, these semiconducting photocatalysts harness light energy to generate electron–hole pairs, which subsequently lead to the formation of reactive radicals that drive catalytic processes.^{8–10} The potential to use natural light as an energy source is compelling in terms of cost-effectiveness and environmental sustainability. Titanium dioxide (TiO₂) has been extensively explored for its photocatalytic potential,^{11–13} but its wide bandgap (~3.2 eV) limits its photocatalytic efficiency given the limited UV component of sunlight (~4%).¹⁴ Two-dimensional molybdenum disulfide (MoS₂) has attracted attention as a potential alternative due to its narrow direct bandgap of ~1.9 eV, which is well-suited for visible light operation.^{15,16} In contrast to its optically inactive bulk form, MoS₂ in the form of single-layer nanosheets exhibits a high density of exposed active edge sites, which enhances its photocatalytic capabilities.^{17,18} In addition, its high specific surface

area and fast charge carrier dynamics contribute to high catalytic efficiency, thereby enhancing its suitability as a novel photocatalyst with broadened application potential.^{19–22} To date, however, it remains difficult to use MoS₂ readily as a photocatalyst. This is primarily due to its limited stability when dispersed in aqueous or organic media, caused by strong attractive van der Waals interactions between nanosheets.²³ Surface chemical modification and the use of select surface active agents can produce relatively stable dispersions under controlled conditions,^{24–26} but these systems are generally not robust against the presence of other species, and are therefore not amenable for general purpose photocatalysis.

Immobilization of MoS₂ nanosheets on suitable supports presents an alternative approach for presenting single- and few-layer MoS₂ active sites to advance the chemical transformations of interest using photocatalysis. Nanocomposites of MoS₂ nanosheets have been successfully prepared in different contexts, including using polymeric,^{27,28} liquid crystal,²⁹ and hydrogel matrices.^{30,31} For the purpose of advancing photocatalysis in solution, we desire a system in which MoS₂ nanosheets can be immobilized in discrete yet porous solid materials. The porous structure would facilitate access of species in solution to the catalyst particles, allowing catalysis to occur effectively.^{32,33} Moreover, the ability to prepare porous solid materials on colloidal length scales, and potentially with magnetic inclusions, offers the possibility of facile catalyst recovery by centrifugation, filtration, or the use of magnetic methods.³⁴ Taking advantage of these characteristics, various studies have described the immobilization of UV photocatalysts, such as TiO₂,^{35,36} copper oxide (CuO),^{37,38} zinc oxide

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(ZnO),^{39,40} and graphitic carbon nitride (g-CN),^{41,42} on porous supports. Although there are several previous studies related to MoS₂-based photocatalysts (Table S1†),^{43–51} there is a notable scarcity of studies directly addressing the immobilization of MoS₂ nanosheets and the visible light photocatalysis thereby enabled.

Here, we demonstrate a successful immobilization strategy for MoS₂ nanosheets. We leverage the concept of microporous supports that effectively mitigate the agglomeration of MoS₂ through physical anchoring. We conducted a comprehensive analysis of the dispersion state of MoS₂ within porous beads through electron microscopy observations and Raman analysis. We demonstrate that the porous bead-supported MoS₂ catalysts exhibit excellent activity for visible light photodegradation of model compounds, while allowing readily retrieval and reuse.

Results and discussion

Design of the MoS₂-immobilized porous bead photocatalyst

MoS₂ has a crystal structure consisting of weakly coupled layers of S–Mo–S, where a Mo atom layer is sandwiched between two layers of S atoms (Fig. S1a†). These layers are held together by weak van der Waals interactions that drive stacking.⁵² To exploit MoS₂ as a photocatalyst, chemically active edge sites must be exposed by exfoliation into nanosheets. Maintaining high catalytic activity requires that the nanosheet morphology is preserved, which necessitates suppressing random aggregation and/or restacking of the nanosheets. We propose a microbead photocatalyst system in which MoS₂ nanosheets are immobilized in alginate-based porous microbeads, as illustrated in Fig. 1. First, MoS₂ nanosheets were obtained by chemical exfoliation using *n*-butyllithium using previously reported methods.⁵³ Briefly, bulk MoS₂ powder was subjected to a reaction with *n*-butyllithium in hexane, yielding Li-intercalated MoS₂. Subsequently, this compound was hydrolyzed in water, resulting in the formation of a stable suspension comprising discrete single- and few-layer nanosheets. The obtained MoS₂ nanosheets were dispersed in sodium alginate solution through ultrasonic dispersion to prepare a precursor solution. The precursor solution was extruded through a

syringe needle into a CaCl₂ solution, leading to the formation of MoS₂-immobilized porous beads (MPBs) (Fig. 1a). When this precursor solution comes into contact with the CaCl₂ medium, ionic crosslinking between the G units of alginate and Ca²⁺ ions is initiated at the surface and gradually progresses toward the center of the bead, eventually solidifying the entire construct. These MPBs are used for photodegradation of a model organic compound, MO, under visible light, as schematically illustrated in Fig. 1b.

Characterization of MoS₂ nanosheets inside microbeads

We first investigated the morphology of the chemically exfoliated MoS₂ nanosheets. The MoS₂ nanosheets exhibited an average width of ~158 nm and an average thickness of ~1.2 nm (Fig. 2a and S2†); the average thickness is consistent with a combination of single- (0.65 nm thickness) and few-layer nanosheets. Chemical exfoliation leads to a mixture of metallic (1T) and semiconducting (2H) nanosheets.⁵⁴ We used

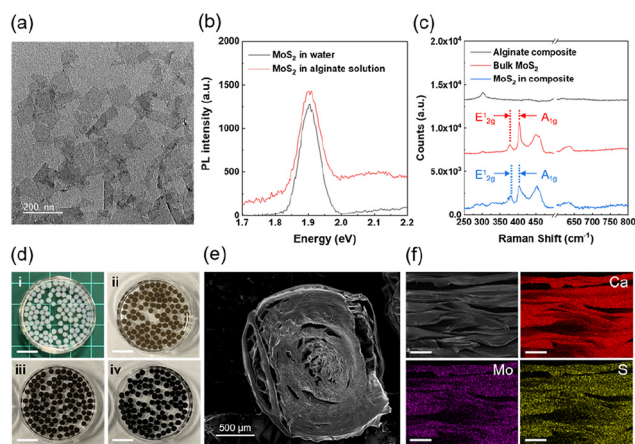


Fig. 2 (a) TEM image of chemically exfoliated MoS₂ nanosheets. (b) Photoluminescence spectra of MoS₂ dispersion. (c) Raman spectra of the alginate composite, bulk MoS₂ and MPBs. (d) Photographs of (i) alginate porous beads (PBs) and MPBs with varying concentrations of incorporated MoS₂: (ii) 0.1, (iii) 0.2, and (iv) 0.5 mg g⁻¹, respectively (scale bar = 10 mm). (e) Cross-section SEM image of the MPBs. (f) Magnified internal morphology of the MPBs and EDX mapping of the MoS₂ nanosheets inside the microbeads (scale bar = 100 μm).

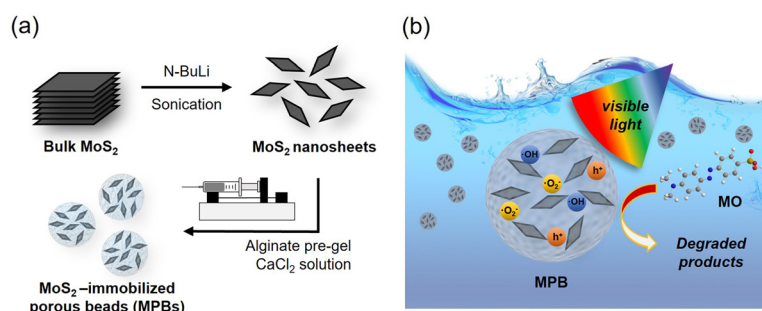


Fig. 1 Schematic illustration for (a) the fabrication of MoS₂-immobilized porous microbeads (MPBs) and (b) the visible-light-induced photocatalytic performance of MPBs.



the optoelectronic properties of the semiconducting fraction for diagnostic purposes. Photoluminescence (PL) spectra confirm the presence of the 2H semiconducting phase contained in these chemically exfoliated MoS₂ nanosheets. As shown in Fig. 2b, the MoS₂ aqueous dispersion showed a sharp peak at 1.9 eV, which is consistent with the bandgap energy of single-layer MoS₂. The PL emission was maintained when dispersed in alginate solution. Using Raman spectroscopy, we further confirmed that the MoS₂ exists as nanosheets within the Ca²⁺-alginate composite (Fig. 2c) by analyzing the wavenumber difference (Δk) between the E_{12g} and A_{1g} vibrational modes.⁵⁵ MoS₂ nanosheets dispersed in the polymer beads had a Δk value of $22.3 \pm 0.3 \text{ cm}^{-1}$, which is notably smaller than the value of $25.9 \pm 0.6 \text{ cm}^{-1}$ observed for bulk MoS₂ powder (Fig. 2c) and consistent with the preservation of exfoliated nanosheet structures in the polymer beads.

MPBs created *via* a conventional extrusion dripping method displayed a spherical morphology (Fig. 2d). Employing a syringe pump facilitated the production of uniform beads with an average diameter of approximately 2.7 mm (Fig. S3†). The size of the beads was adjustable based on the inner diameter of the needle (Fig. S4†). We prepared MPBs using different concentrations of MoS₂, denoted as MPB_{*n*}, where *n* represents the MoS₂ concentration (mg g⁻¹). The concentration of added MoS₂ affected the color of the beads, from dark brown to black, but had no noticeable effect on their size. However, it is noteworthy that the Δk value for beads with the largest MoS₂ concentration, MPB_{0.5}, exhibited a minor increase compared to that of MPB_{0.2} (Fig. S5†). This suggests that the propensity for nanosheet restacking increases at elevated mass loadings. Fig. 2e shows the internal morphology of the MPBs, whose structure is characterized by pores formed by a dense alginate-calcium layer. The surface of the alginate-Ca layer without MoS₂ is smooth, whereas the surface containing MoS₂ is relatively rough (Fig. S6a-c†). For MPB_{0.1} and MPB_{0.2}, MoS₂ nanosheets are well dispersed within the alginate-Ca layer. Elemental dispersive X-ray spectroscopy (EDS) mapping indicates visually that the distribution of MoS₂ nanosheets within MPB_{0.2} is spatially uniform (Fig. 2f). However, in MPB_{0.5}, some nanosheets remain well anchored, while others exhibit restacking and aggregation (Fig. S6d†). These observations are consistent with the Raman spectroscopy data and are expected to have a significant impact on photocatalytic performance.

Photocatalytic performance of MPBs under visible light

We assessed the efficacy of MPBs as a “green” photocatalytic system for the photodegradation of molecular species. Methyl orange (MO) was chosen as a model compound, given its well-known role as an environmental pollutant.⁵⁶ The photocatalytic degradation of MO was initiated by the introduction of MPBs into an aqueous MO solution, followed by stirring under visible light generated by a custom-made cylindrical photoreactor enveloped by an LED strip (Fig. S7†). To avoid the sedimentation of MPBs, which may reduce the effective surface area for light absorption, mild stirring is required. MO exhibits distinctive colors depending on the pH of its environ-

ment: red under acidic conditions and yellow in neutral and basic environments. The investigation of MO photodegradation involved UV-visible spectroscopy, with absorbance intensity monitored at a wavelength of 500 nm in the acidic solution and 464 nm under neutral and basic conditions, respectively (Fig. S8†). The adsorption-desorption equilibrium of MO was reached in 30 minutes in a dark environment (Fig. S9a†). The amount of MO adsorbed onto the beads decreased with increasing pH. This is attributed to the fact that MO exhibits a negative charge above pH 7,⁵⁷ rendering it less amenable to adsorption by the alginate beads, which also possess a negative charge. Photolysis of MO was negligible when subjected to visible light in the absence of the beads (Fig. S9b†). In the presence of the beads, as shown in Fig. 3a-c, there was no noticeable photodegradation of MO under neutral and basic conditions, whereas under acidic conditions, a significant decrease in absorbance was observed as a function of irradiation time. This means that an active photocatalytic reaction occurs, which is further evidenced by the loss of color of the MO solution.

This acidic condition-favorable photodegradation of MO can be understood through the photocatalytic mechanism of MoS₂.⁴⁶ Semiconducting MoS₂ nanosheets generate electron-hole pairs when exposed to visible light (1). These photogenerated electrons participate in reactions with dissolved oxygen to form superoxide radicals ([•]O₂⁻) (2). Subsequently, [•]O₂⁻ can further react with protons (H⁺) to generate hydrogen peroxide (H₂O₂) species (3). The ensuing decomposition of H₂O₂ results in the formation of hydroxyl radicals ([•]OH) (4). These hydroxyl

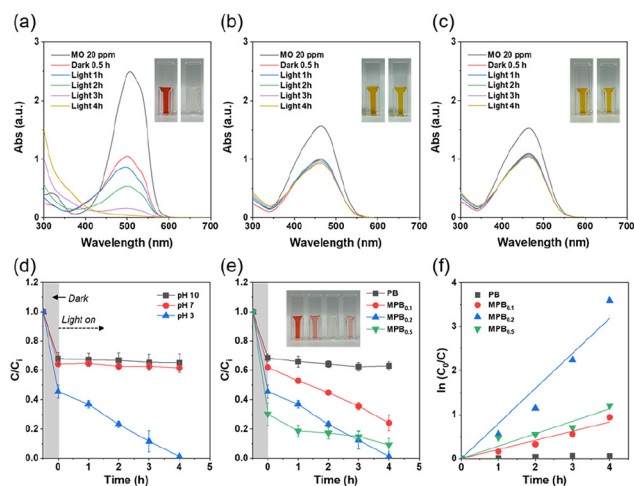
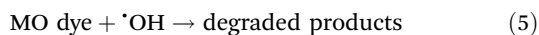
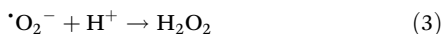


Fig. 3 UV-vis spectra of methyl orange (MO) during photodegradation with the MPB catalyst at (a) pH 3, (b) pH 7, and (c) pH 10, respectively. The insets show the photographs of MO solution before (left) and after (right) photodegradation. Plots of C/C_0 versus time depending on the reaction conditions: (d) pH and (e) MoS₂ concentration. C_0 is the initial concentration of MO and C is the concentration of MO at reaction time t . The inset of (e) shows the photograph of MO solution after the photodegradation reaction (from left to right: PB, MPB_{0.1}, MPB_{0.2}, and MPB_{0.5}). (f) Plots of $\ln(C_0/C)$ versus time for the photodegradation of MO using different catalysts at pH 3.



radicals play a pivotal role in the degradation of organic species (5), with $\cdot\text{O}_2^-$ also contributing, though to a lesser extent.^{48,58,59}



Within this sequence of processes, the presence of H^+ serves to facilitate the generation of hydroxyl radicals. The observed pH dependence (Fig. 3d) in the photodegradation of MO by the bead-supported MoS_2 nanosheets is consistent with this proton facilitated mechanism.

The photocatalytic efficiency of MPBs is affected by several parameters, including bead size, the degree of crosslinking, and the concentration of MoS_2 in the beads. Larger bead sizes and lower degrees of crosslinking tend to yield slightly reduced degradation rates under visible light of 92.7% and 82.3% after 4 h, respectively (Fig. S10†). In general, the smaller the bead, the larger the specific surface area, leading to better photocatalytic efficiency. With respect to the degree of crosslinking, we speculate that higher levels of crosslinking result in better preservation of nanosheet dispersion during bead formation. Independent of photocatalytic degradation, the incorporation of MoS_2 nanosheets increases the adsorption capacity of the beads for MO, as revealed by experiments conducted in the dark (Fig. 3e). The quantity of MO adsorbed by the beads (represented by the initial decline in the MO concentration in the dark) increased steadily with the concentration of the MoS_2 suspension used to prepare the beads. The effectiveness of the photodegradation increased as the concentration of MoS_2 used to prepare the beads increased from 0.1 to 0.2 mg g^{-1} . At 0.5 mg g^{-1} , however, the rate of photodegradation decreased. Although the difference in absolute values is minor, the observed increase in Δk values at this higher concentration is likely attributed to the restacking of the nanosheets, as corroborated by the Raman spectroscopy data (Fig. S5†). We presume that the photodegradation efficacy is a function of the quantity of semiconducting vs. metallic MoS_2 nanosheets present in the system. The fraction of semiconducting nanosheets is unknown, but their prevalence can be increased through thermal annealing,⁶⁰ which we conducted here to investigate its effect on photodegradation. As seen from the UV-Vis data (Fig. S11a†), this enables the characteristic excitonic absorption peaks at 605 and 648 nm to be readily distinguished from the background absorption. However, the increased tendency of semiconducting sheets to restack limits the use of higher concentrations in bead preparation (Fig. S11b†). Regarding the reaction kinetics, the MPB catalyst exhibits behavior consistent with pseudo-first-order kinetics (Fig. 3f). The degradation rate constant (k) can be

deduced from the equation $\ln(C_0/C) = kt$, where C_0 represents the initial concentration of MO after adsorption, C signifies the concentration of MO, and t denotes the reaction time. The calculated degradation rate constants for MO in conjunction with MPBs were determined to be 0.21 h^{-1} for $\text{MPB}_{0.1}$, 0.80 h^{-1} for $\text{MPB}_{0.2}$, and 0.28 h^{-1} for $\text{MPB}_{0.5}$, respectively.

Recoverable and reusable microbead photocatalyst

We assessed the reusability of the MPB catalyst. Following its initial use, the beads were collected by centrifugation. After a gentle rinse with DI water, the catalyst was prepared for the subsequent reaction. For each reaction cycle, the establishment of an adsorption-desorption equilibrium for MO was initiated in the dark for 30 minutes before the operation of the visible light photoreactor. The adsorption amount, which was close to 60% in the first cycle, decreased to about 25% in the next cycle, probably due to residual MO degraded products adsorbed within the beads from the last batch (Fig. 4a). Nevertheless, the degradation efficiency remained consistently high at 99% even after four cycles. In the context of the C/C_i plot, a relatively modest slope was evident within the first hour following visible light irradiation during the first cycle, primarily attributable to the preferential degradation of MO adsorbed within the beads (Fig. 4b). Thereafter, similar slopes were consistently observed throughout the subsequent cycles, reflecting the kinetic trends observed in all four reuse cases. The MPB catalyst demonstrated consistent performance without significant discernible physical deterioration (Fig. 4c). This excellent performance is comparable to existing MoS_2 -based catalysts and is especially noteworthy in terms of the economical catalyst amount, mild light intensity, and easy reuse (Table S1†). These findings imply that our MPB catalysts have great potential as high-performance green catalyst platforms.

Experimental

Materials

Sodium alginate (SA, from brown algae), calcium chloride (CaCl_2 , anhydrous, >97%), molybdenum(IV) sulfide (MoS_2 , powder, 99%),

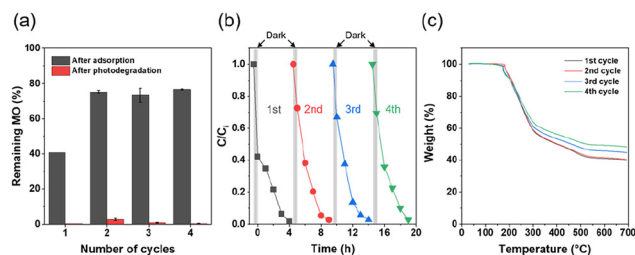


Fig. 4 Recycling properties of the photocatalytic degradation of MO over the MPB catalyst. (a) Adsorption and photodegradation efficiency for each cycle. The remaining MO is calculated as $C/C_i \times 100$ (%), where C_i is the initial MO concentration and C is the concentration of MO after adsorption and photodegradation, respectively. (b) Plots of C/C_i versus time for each cycle. (c) TGA curves of MPBs after each cycle.



particle size $<2\ \mu\text{m}$), *n*-butyllithium solution (*n*-BuLi, 2.5 M in hexane), and hexane (anhydrous, 95%) were purchased from Sigma-Aldrich (USA). Methyl orange (MO), sodium hydroxide solution (1N), and hydrochloric acid solution (1N) were purchased from Fisher Chemical (USA). Deionized double-distilled (DI) water was used in all experiments.

Preparation of MoS₂ nanosheets

2D MoS₂ nanosheets were prepared *via* a typical chemical exfoliation procedure. Initially, bulk MoS₂ powder was introduced into an *n*-BuLi solution at a molar ratio of 1 : 3 in an N₂-filled glovebox. This mixture was allowed to stir for 48 hours under an N₂ atmosphere. Subsequently, it underwent several cycles of washing with hexane and centrifugation at 6000 rpm for 30 minutes to separate excess *n*-BuLi from the Li-intercalated MoS₂. This washing process was repeated three times. Then, the residual hexane was evaporated under vacuum. Degassed water was then added to the resulting pellet. Subsequently, the solution was bath sonicated for 90 minutes to promote the exfoliation of the Li-intercalated 2D nanosheets. Following sonication, the unexfoliated material was isolated through centrifugation at 2000 rpm for 10 minutes, a step that was repeated twice. The dark supernatant obtained was subsequently transferred into dialysis bags with a molecular weight cutoff of 20 kDa (Sigma) for dialysis in DI water over a period of 3 days. The concentration of the final MoS₂ suspension was determined by weighing the restacked nanosheets on a glass coverslip after the water had evaporated.

Fabrication of MPBs

We fabricated MPBs using an extrusion dripping method. The initial step involved the preparation of a precursor solution, in which MoS₂ nanosheets were dispersed in SA aqueous solution. To achieve this, SA was completely dissolved in DI water. Subsequently, the MoS₂ nanosheet solution was introduced into the prepared SA solution, followed by probe sonication (Q700, QSonica, USA) at 60% amplitude to ensure homogeneity. The concentration of SA was set to 5 wt%, while the concentration of MoS₂ varied at 0.1, 0.2, and 0.5 mg g⁻¹ within the entire solution, denoted as MPB_{0.1}, MPB_{0.2}, and MPB_{0.5}, respectively. The resulting precursor solutions were extruded into CaCl₂ solution (5 wt%) dropwise using a syringe equipped with a needle (25 G) under mild stirring. For uniform bead fabrication, a syringe pump (KDS 210, KD Scientific, USA) was employed, and the flow rate was set to 0.2 ml min⁻¹. The Ca²⁺ ion crosslinking proceeded for 12 hours at room temperature. Finally, MPBs were successfully obtained. The MPBs were rinsed using DI water to wash the excess Ca²⁺ ions.

Characterization of MPBs

The morphology of the MoS₂ nanosheets was observed with a transmission electron microscope (TEM, JEM-F200 F2, JEOL, USA) operating at 200 kV. The thickness of the MoS₂ nanosheets was measured from an atomic force microscopy (AFM, Dimension Icon, Bruker, USA) image. PL spectra were collected using an Edinburgh Instruments FLS1000 fluorom-

eter with an excitation light of 532 nm. Raman spectra were obtained using a Horiba LabRam Evolution confocal Raman microscope with 633 nm light excitation and 300 s exposure under a 100× objective. The average diameter of MPBs was determined using ImageJ software. The morphology of freeze-dried MPBs and the distribution of MoS₂ nanosheets were analyzed by environmental scanning electron microscopy (SEM, FEI Quanta 600, FEI, USA) coupled with energy-dispersive X-ray spectroscopy (EDX). Thermogravimetric analysis (TGA) was carried out using an SDT Q650 (TA Instruments, USA) at a scan rate of 10 °C min⁻¹.

Photocatalytic performance test

To confirm the photocatalytic performance of the MPB catalyst, MO was employed as a representative reaction model. In a typical experimental setup, 10 mL of MO aqueous solution was prepared in a glass vial at a concentration of 20 ppm (20 mg L⁻¹). Subsequently, the pH of the solution was adjusted to 3, 7, and 10, respectively, utilizing HCl and NaOH solutions. MPBs were then introduced into each of these solutions. As a visible light source, a 24 W m⁻¹ LED strip was utilized. The vial containing the MPB and MO solutions was placed inside a custom-made cylindrical reactor enveloped by a 5-meter LED strip (Fig. S7†). To mitigate temperature increases resulting from LED illumination, a USB-powered fan was positioned above the reactor, and an additional fan was placed adjacent to the reactor to facilitate continuous cooling. Before the light irradiation, the vial was shielded with foil and stirred for 30 minutes in the dark to establish an adsorption–desorption equilibrium of MO. The photodegradation process was subsequently initiated with continuous stirring. The degradation progress was tracked by recording UV–visible spectra at designated time intervals using a Varian Cary 100 Bio UV–visible spectrometer.

Conclusions

In summary, we described here a method to support MoS₂ nanosheets within porous polymeric microbeads, ensuring their high dispersion stability. The porous network of the beads serves as a physical scaffold that prevents aggregation and restacking of the MoS₂ nanosheets. As a result, the photocatalytic activity of the nanosheets is preserved through several cycles of use. The ability to use visible light to induce photocatalytic degradation, or other carrier-driven processes, and the ease with which the beads can be retrieved and reused stand out as attractive properties. These results highlight the potential of these polymeric microbeads for the development of innovative green and sustainable technologies for water treatment or other carrier-driven chemical processes.

Author contributions

This manuscript was written through the contributions of all authors. D. Park: conceptualization, methodology, formal ana-



lysis, and writing. J. W. Kim: writing and funding acquisition. C. O. Osuji: conceptualization, writing, supervision, and project administration.

Data availability

The data supporting this article have been included as part of the ESI.†

Conflicts of interest

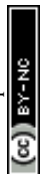
There are no conflicts of interest to declare.

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References

- 1 A. Rafiq, M. Ikram, S. Ali, F. Niaz, M. Khan, Q. Khan and M. Maqbool, *J. Ind. Eng. Chem.*, 2021, **97**, 111–128.
- 2 M. Saeed, M. Muneer, A. U. Haq and N. Akram, *Environ. Sci. Pollut. Res. Int.*, 2022, **29**, 293–311.
- 3 P. Sun, Z. Xing, Z. Li and W. Zhou, *Chem. Eng. J.*, 2023, **458**, 141399.
- 4 C. Feng, Z. Lu, Y. Zhang, Q. Liang, M. Zhou, X. Li, C. Yao, Z. Li and S. Xu, *Chem. Eng. J.*, 2022, **435**, 134833.
- 5 E. M. Bayan, L. E. Pustovaya and M. G. Volkova, *Environ. Technol. Innovation*, 2021, **24**, 101822.
- 6 J. Zeng, Z. Li, H. Jiang and X. Wang, *Mater. Horiz.*, 2021, **8**, 2964–3008.
- 7 Z. Zhou, B. Li, X. Liu, Z. Li, S. Zhu, Y. Liang, Z. Cui and S. Wu, *ACS Appl. Bio Mater.*, 2021, **4**, 3909–3936.
- 8 X. Xiao, S. Tu, M. Lu, H. Zhong, C. Zheng, X. Zuo and J. Nan, *Appl. Catal., B*, 2016, **198**, 124–132.
- 9 N. Serpone and A. V. Emeline, *J. Phys. Chem. Lett.*, 2012, **3**, 673–677.
- 10 M.-H. Wu, L. Li, N. Liu, D.-J. Wang, Y.-C. Xue and L. Tang, *Process Saf. Environ. Prot.*, 2018, **118**, 40–58.
- 11 Q. Guo, C. Zhou, Z. Ma and X. Yang, *Adv. Mater.*, 2019, **31**, e1901997.
- 12 M. Ni, M. K. H. Leung, D. Y. C. Leung and K. Sumathy, *Renewable Sustainable Energy Rev.*, 2007, **11**, 401–425.
- 13 C. B. Anucha, I. Altin, E. Bacaksiz and V. N. Stathopoulos, *Chem. Eng. J. Adv.*, 2022, **10**, 100262.
- 14 T. Larbi, M. A. Amara, B. Ouni and M. Amlouk, *Mater. Res. Bull.*, 2017, **95**, 152–162.
- 15 Z. Li, X. Meng and Z. Zhang, *J. Photochem. Photobiol., C*, 2018, **35**, 39–55.
- 16 E. Parzinger, B. Miller, B. Blaschke, J. A. Garrido, J. W. Ager, A. Holleitner and U. Wurstbauer, *ACS Nano*, 2015, **9**, 11302–11309.
- 17 M.-H. Wu, L. Li, Y.-C. Xue, G. Xu, L. Tang, N. Liu and W.-Y. Huang, *Appl. Catal., B*, 2018, **228**, 103–112.
- 18 W.-Q. Chen, L.-Y. Li, L. Li, W.-H. Qiu, L. Tang, L. Xu, K.-J. Xu and M.-H. Wu, *Engineering*, 2019, **5**, 755–767.
- 19 A. Rahman, J. R. Jennings, A. L. Tan and M. M. Khan, *ACS Omega*, 2022, **7**, 22089–22110.
- 20 S. Huang, C. Chen, H. Tsai, J. Shaya and C. Lu, *Sep. Purif. Technol.*, 2018, **197**, 147–155.
- 21 H. Shen, S. Liao, C. Jiang, J. Zhang, Q. Wei, R. A. Ghiladi and Q. Wang, *Carbohydr. Polym.*, 2022, **277**, 118853.
- 22 Y. Yu, L. Lu, Q. Yang, A. Zupanic, Q. Xu and L. Jiang, *ACS Appl. Nano Mater.*, 2021, **4**, 7523–7537.
- 23 M. Chhowalla, H. S. Shin, G. Eda, L. J. Li, K. P. Loh and H. Zhang, *Nat. Chem.*, 2013, **5**, 263–275.
- 24 W. Osim, A. Stojanovic, J. Akbarzadeh, H. Peterlik and W. H. Binder, *Chem. Commun.*, 2013, **49**, 9311–9313.
- 25 D. M. Sim, H. J. Han, S. Yim, M. J. Choi, J. Jeon and Y. S. Jung, *ACS Omega*, 2017, **2**, 4678–4687.
- 26 K. Zhou, J. Liu, P. Wen, Y. Hu and Z. Gui, *Appl. Surf. Sci.*, 2014, **316**, 237–244.
- 27 D. Xuan, Y. Zhou, W. Nie and P. Chen, *Carbohydr. Polym.*, 2017, **155**, 40–48.
- 28 X. Feng, X. Wang, W. Xing, K. Zhou, L. Song and Y. Hu, *Compos. Sci. Technol.*, 2014, **93**, 76–82.
- 29 U. R. Gabinet, C. Lee, R. Poling-Skutvik, D. Keane, N. K. Kim, R. Dong, Z. Vicars, Y. Cai, A. U. Thosar, A. Grun, S. M. Thompson, A. J. Patel, C. R. Kagan, R. J. Composto and C. O. Osuji, *ACS Mater. Lett.*, 2021, **3**, 704–712.
- 30 H. P. Lee, G. Lokhande, K. A. Singh, M. K. Jaiswal, S. Rajput and A. K. Gaharwar, *Adv. Mater.*, 2021, **33**, e2101238.
- 31 W. Xu, W. Wang, S. Chen, R. Zhang, Y. Wang, Q. Zhang, L. Yuwen, W. J. Yang and L. Wang, *J. Colloid Interface Sci.*, 2021, **586**, 601–612.
- 32 M. Xia, S.-M. Kang, G.-W. Lee, Y. S. Huh and B. J. Park, *J. Ind. Eng. Chem.*, 2019, **73**, 306–315.
- 33 L. Ai, H. Yue and J. Jiang, *J. Mater. Chem.*, 2012, **22**, 23447–23454.
- 34 D. Park, C. O. Osuji and J. W. Kim, *Small Methods*, 2023, **7**, e2201195.
- 35 N. X. D. Mai, J. Bae, I. T. Kim, S. H. Park, G.-W. Lee, J. H. Kim, D. Lee, H. B. Son, Y.-C. Lee and J. Hur, *Environ. Sci.: Nano*, 2017, **4**, 955–966.
- 36 S. Ullah, E. P. Ferreira-Neto, A. A. Pasa, C. C. J. Alcântara, J. J. S. Acuña, S. A. Bilmes, M. L. Martínez Ricci, R. Landers, T. Z. Fermino and U. P. Rodrigues-Filho, *Appl. Catal., B*, 2015, **179**, 333–343.
- 37 S. Sharma and S. Basu, *Sep. Purif. Technol.*, 2021, **279**, 119759.
- 38 M. Diab, K. Shreth, M. Volokh and T. Mokari, *Molecules*, 2021, **26**, 6067.



- 39 K. O. Kassem, M. A. T. Hussein, M. M. Motawea, H. Gomaa, Z. A. Alrowaili and M. Ezzeldien, *J. Cleaner Prod.*, 2021, **326**, 129416.
- 40 M. Hasanpour, S. Motahari, D. Jing and M. Hatami, *Top. Catal.*, 2024, **67**, 1334–1347.
- 41 Q. Cao, J. Barrio, M. Antonietti, B. Kumru, M. Shalom and B. V. K. J. Schmidt, *ACS Appl. Polym. Mater.*, 2020, **2**, 3346–3354.
- 42 X. Zhang, H. Zhang, P. Chen, M. Liu, P. Wu, C. Liu and W. Jiang, *Colloids Surf., A*, 2022, **644**, 128814.
- 43 T.-M. Tien, C.-H. Chen, C.-T. Huang and E. L. Chen, *Catalysts*, 2022, **12**, 1474.
- 44 J. Kisała, R. Wojnarowska-Nowak and Y. Bobitski, *Sci. Rep.*, 2023, **13**, 14148.
- 45 X. Lin, X. Wang, Q. Zhou, C. Wen, S. Su, J. Xiang, P. Cheng, X. Hu, Y. Li, X. Wang, X. Gao, R. Nözel, G. Zhou, Z. Zhang and J. Liu, *ACS Sustainable Chem. Eng.*, 2018, **7**, 1673–1682.
- 46 W. Liu, Q. Hu, F. Mo, J. Hu, Y. Feng, H. Tang, H. Ye and S. Miao, *J. Mol. Catal. A: Chem.*, 2014, **395**, 322–328.
- 47 W. Zhang, X. Xiao, L. Zheng and C. Wan, *Appl. Surf. Sci.*, 2015, **358**, 468–478.
- 48 H. K. Sadhanala, S. Senapati, K. V. Harika, K. K. Nanda and A. Gedanken, *New J. Chem.*, 2018, **42**, 14318–14324.
- 49 D. Sahoo, J. Shakya, N. Ali, W. J. Yoo and B. Kaviraj, *Langmuir*, 2022, **38**, 1578–1588.
- 50 D. Sahoo, B. Kumar, J. Sinha, S. Ghosh, S. S. Roy and B. Kaviraj, *Sci. Rep.*, 2020, **10**, 10759.
- 51 Z. Xia, Y. Tao, Z. Pan and X. Shen, *Results Phys.*, 2019, **12**, 2218–2224.
- 52 N. Thomas, S. Mathew, K. M. Nair, K. O'Dowd, P. Forouzandeh, A. Goswami, G. McGranaghan and S. C. Pillai, *Mater. Today Sustainability*, 2021, **13**, 100073.
- 53 X. Lu, U. R. Gabinet, C. L. Ritt, X. Feng, A. Deshmukh, K. Kawabata, M. Kaneda, S. M. Hashmi, C. O. Osuji and M. Elimelech, *Environ. Sci. Technol.*, 2020, **54**, 9640–9651.
- 54 S. Jayabal, J. Wu, J. Chen, D. Geng and X. Meng, *Mater. Today Energy*, 2018, **10**, 264–279.
- 55 H. Li, Q. Zhang, C. C. R. Yap, B. K. Tay, T. H. T. Edwin, A. Olivier and D. Baillargeat, *Adv. Funct. Mater.*, 2012, **22**, 1385–1390.
- 56 M. Farhan Hanafi and N. Sapawe, *Mater. Today: Proc.*, 2020, **31**, A141–A150.
- 57 E. Akama, A. J. Tong, M. Ito and S. Tanaka, *Talanta*, 1999, **48**, 1133–1137.
- 58 Z. Zhang, G. Wang, W. Li, L. Zhang, T. Chen and L. Ding, *Colloids Surf., A*, 2020, **601**, 125034.
- 59 A. P. Nagvenkar and A. Gedanken, *ACS Appl. Mater. Interfaces*, 2016, **8**, 22301–22308.
- 60 Z. Wang, Y. J. Zhang, M. Liu, A. Peterson and R. H. Hurt, *Nanoscale*, 2017, **9**, 5398–5403.

