Nanoscale Horizons



View Article Online REVIEW



Cite this: Nanoscale Horiz., 2025, **10**, 2304

Synthesis planning for atomically precise metal nanoclusters

The rational design and synthesis of materials with tailored properties remains a long-standing goal in advanced materials science. Metal nanoclusters (MNCs), distinguished by their atomic precision and molecule-like properties-including discrete energy levels, strong photoluminescence, and high property tunability-represent promising platforms for applications spanning catalysis to biomedicine. This perspective presents a comprehensive synthesis planning framework comprising three critical stages, i.e., target design, route development, and condition optimization, systematically addressing MNC rational design and synthesis with special emphasis on thiolate-protected gold nanoclusters as exemplary systems. We first discuss design considerations for core and ligand shell engineering based on their profound influence on overall material properties. Subsequently, we examine methods and synthetic mechanisms for atomic-level tailoring of core and ligand shells to achieve target MNC synthesis. We then elucidate condition parameter tuning considerations based on their deterministic roles in reaction outcomes. While this structured approach provides a systematic methodology for MNC development, significant challenges persist owing to the high structural and synthetic complexity of MNCs. We then discuss the opportunities brought by recent advances in machine learning and high-throughput experimentation, which have demonstrated potential in addressing these challenges based on their superior computational and data analytical capabilities. We advocate for systematic adoption of this synthesis planning approach enhanced by data-driven methods, addressing inherent limitations in future development to better exploit these integrated approaches for accelerating rational MNC design and synthesis.

Received 18th May 2025, Accepted 2nd July 2025

DOI: 10.1039/d5nh00353a

rsc li/nanoscale-horizons

^a Joint School of National University of Singapore and Tianjin University, International Campus of Tianjin University, Fuzhou 350207, P. R. China. E-mail: chexiej@nus.edu.sg

1. Introduction

Rational design and synthesis represents the strategic process by which scientists systematically design and execute the efficient synthesis of desired compounds with specific properties. In contrast to traditional trial-and-error approaches that



Jingkuan Lyu

Jingkuan Lyu is a PhD student at the National University of Singapore under the supervision of Prof. Jianping Xie. He obtained his BS degree in chemical engineering from the same university. His research interest focuses on theautomated synthesis of gold nanoclusters, integrating in situ and real-time characterization, robotic synthesis and machine learning.



Jing Qian

Jing Qian obtained his BS (2021) from the National University of Singapore. He is currently a PhD student under the supervision of Prof. Jianping Xie at NUS. His current research interests are centered on thestructure prediction of metal nanoclusters with the assistance of machine learning algorithms.

^b Department of Chemical and Biomolecular Engineering, National University of Singapore, Singapore 117585, Singapore

rely on serendipitous discovery, this methodology begins with identifying target materials possessing specific desired characteristics, guided by established structure-property relationships. Synthetic mechanistic insights are then leveraged to develop viable pathways between available precursors and the target products-exemplified by retrosynthetic analysis in organic chemistry. The field of drug discovery has adopted the design-make-test-analyse (DMTA) framework to implement this systematic approach, embracing the iterative nature of rational design through continuous validation and refinement of design decisions via experimental feedback. Crucially, data generated throughout these cycles not only inform immediate synthetic objectives but also enrich the broader knowledge base, creating a cumulative foundation for increasingly sophisticated materials development strategies.

Atomically precise metal nanoclusters (MNCs) represent a unique class of nanomaterials distinguished by their monodispersity and well-defined molecular structures. These ultrasmall entities consist of a metallic core, typically below 3 nm in diameter, stabilized by a shell of metal-ligand motifs. Singlecrystal X-ray diffraction (SCXRD) and mass spectrometry characterization enable precise determination of their structure and composition, often expressed in the form $[M_n(L)_m]^q$, where n and m denote the number of metal atoms (M) and ligands (L) respectively, and q is the net charge. Owing to their quantum confinement effects—arising when cluster dimensions become comparable to the electron de Broglie wavelength-MNCs exhibit molecule-like behaviours such as electron transition between the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO),2 strong photoluminescence (PL),³⁻⁷ and intrinsic chirality,^{8,9} rendering them promising in diverse applications including catalysis, 10-12 biomedicine, 13-15 and photosensitizers. 16

The molecule-like character renders MNC properties highly reliant on the cluster's size, structure, and composition, all engineerable via atomic-level manipulation of the metallic core and ligand shell.17-22 For instance, the growth of the biicosahedral metal core in $[Au_{25}(PR_3)_{10}(SR)_{10}Cl_2]^+$ to a triicosahedral structure in [Au₃₇(PR₃)₁₀(SR)₁₀Cl₂]⁺ (where PR₃ represents phosphine ligands and SR represents thiolate ligands) significantly reduces the optical energy gap from 1.73 eV to 0.83 eV.²³ In another example, the peroxidase mimic catalytic activity of Au₁₅(SR)₁₃ was enhanced by incorporating ligands with higher electron-withdrawing capabilities.24 Such tunability not only enables fundamental structure-property relationship development, 25-27 but also inspires advances in synthetic methods and growth mechanisms toward stepwise synthesis and control of MNCs, 28,29 as exemplified by the conceptualization of "total synthesis" of MNCs. 30 Understanding MNC formation processes has revealed opportunities for customized MNC synthesis through adjusting synthetic condition parameters, 31-33 such as pH tuning to promote less stable species like highly luminescent Au₂₂(SR)₁₈, 34 or exploiting additive-ligand interactions to control reaction kinetics. 35 However, the chemical complexity, coupled with intricate parameter interdependencies, poses significant challenges to advancing rational design and synthesis towards higher precision. The limited availability of high-quality structural data further exacerbates these difficulties. Recent advances in data-driven methodologies-integrating machine learning (ML) algorithms with highthroughput experimentation (HTE)—have demonstrated promising capabilities in modelling complex relationships for rational design and synthesis across diverse materials.36-43 Autonomous laboratories exemplify fully integrated systems enabling dataguided material discovery, as demonstrated by Ceder and coworkers, who successfully identified 41 novel compounds. 44 These developments provide timely inspiration for adopting data-centric strategies in MNC research. 45-47

Targeting rational design and synthesis of MNC, this perspective presents a comprehensive synthesis planning



Zhucheng Yang

Zhucheng Yang received his BS (2020) degree from theUniversity of Chinese Academy of Sciences. He is currently pursuing his PhD degree under the supervision of Prof. Jianping Xie at the National University of Singapore. He is interested in the smart synthesis and liganddriven programming of metal nanoclusters.



Jianping Xie

Dr Jianping Xie is a professor at the Department of Chemical & Biomolecular Engineering, National University of Singapore. He received his BS and MS from Tsinghua University and his PhD from the Singapore-MIT Alliance. Since joining NUS in 2010, he has led research on water-soluble noble metal nanoclusters for biomedical and catalytic applications. With 280+ publications, 37 000+ citations, and an Hindex of 105, he has been

recognized as a Highly Cited Researcher (2018-2024, Clarivate). His interests include luminescent metal nanoclusters, nanomedicine, and nanocatalysis. He is an Associate Editor of Aggregate (Wiley).

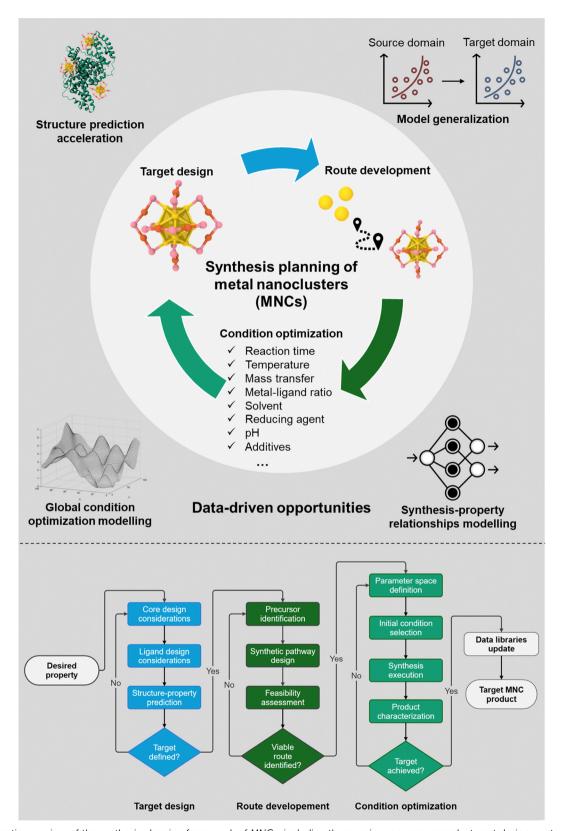


Fig. 1 Schematic overview of the synthesis planning framework of MNCs, including three main processes, namely, target design, route development and condition optimization, and the opportunities endowed by data-driven approach adoption, in terms of structure prediction acceleration, global condition optimization modelling, synthesis-property relationship modelling, and model generalization. Colour label: yellow, core Au or Au cation; orange, motif Au; pink, S. The detailed flowchart below illustrates the individual steps during the three critical stages of synthesis planning for MNCs. After each successful synthesis, all relevant data are updated to the data libraries for future MNC developments, forming an iterative cycle.

framework, which comprises three critical stages: target design (designing the MNC core and ligand shell for desired functionality), route development (mapping the synthetic route from available precursors to targets), and condition optimization (fine-tuning reaction condition parameters for precise synthesis control) (Fig. 1). We emphasize thiolate-protected Au nanoclusters as exemplary systems given their extensive characterization and established relevance. We highlight the challenges faced and discuss how emerging data-driven approaches can address these barriers. Finally, we conclude with our perspectives on how to better exploit this integrated approach in future development. While existing reviews have focused on individual aspects of MNC design and synthesis, 7,19,30,48 no comprehensive framework exists that systematically integrates target design, route development, and condition optimization into a unified synthesis planning approach. Although a recent perspective presents opportunities in MNC synthesis enabled by the integration of automation, data and advanced algorithms, 45 it lacks in-depth discussion of theoretical considerations regarding the synthetic and structural complexities of MNCs. Moreover, previous reviews on data-driven nanomaterial discovery, such as those focusing on nanoparticles, do address the atomic-precision considerations molecular-like property dependencies essential for MNCs. 49,50 By establishing this integrated framework which systematically decomposes MNC rational design and synthesis into tractable relationship modelling problems amenable to data-driven methodologies, this perspective aims to catalyse a paradigm shift toward predictive MNC synthesis that bridges the gap between the precision of molecular design and the complexity of nanomaterial engineering, ultimately enabling more efficient discovery of atomically precise MNCs with tailored properties.

2. Target MNC design

Target design represents the critical first step in MNC synthesis planning, wherein researchers define the structural features needed to achieve specific functional properties prior to synthesis. Different from conventional trial-and-error approaches, rational target design leverages established structure-property relationships to anticipate how microscopic structural features will determine macroscopic characteristics.

For atomically precise MNCs, design efforts focus on two interdependent structural domains: the metallic core and the surrounding ligand shell. Each component plays distinct yet complementary roles in determining the MNC's fundamental properties. Understanding these relationships enables researchers to design MNCs with tailored functionality for specific applications.

The metallic core-defined by its size, structure, and composition—predominantly governs the electronic structure of MNCs.^{2,51} Due to quantum confinement effects at the nanoscale, MNCs exhibit discrete electronic energy levels rather than continuous bands, resulting in molecule-like optical absorption

profiles which are strongly affected by the metallic core.⁵² For instance, Au₄₂(SCH₂Ph)₃₂ (SCH₂Ph represents benzyl mercaptan) with its distinctive rod-like core structure demonstrates strong near-infrared absorption at 808 nm, yielding superior photothermal conversion efficiency compared to more spherical MNC species.⁵³ The core can also significantly influence catalytic activity via size effects, following predictable trends that inform design strategies. Zheng et al. demonstrated that catalytic performance in resazurin reduction systematically increases as Au nanocluster size decreases from Au25 to Au18 to Au₁₅.⁵⁴ Similarly, Li et al. observed enhanced electrochemical CO₂ reduction activity with decreasing Au nanocluster dimensions.⁵⁵ Moreover, variations in core composition can tune MNC properties by modulating the HOMO-LUMO gap, further expanding the design landscape.⁵⁶

While the core determines electronic structure, the protective ligand shell offers additional opportunities for property modulation. In addition to stabilizing the metal core, ligands influence solubility, aggregation behaviour, and interfacial interactions, which can markedly affect properties such as PL, particularly through solvent-mediated effects.⁴ Furthermore, the ligand can also alter MNC properties by exerting electronic effects. For example, replacing electron-donating 3-mercapto-2methylpropanoic acid (MMPA) with N-acetylcysteine (NAC) in Au₁₅(SR)₁₃ nanoclusters enhances the peroxidase-like activity, illustrating the potential for ligand-directed catalytic tuning.²⁴

Thus, rational design of MNCs demands integrated consideration of both core architecture and ligand shell composition to achieve desired property profiles. In the following section, we elaborate on specific principles for templated core design and ligand shell engineering that enable predictive synthesis of MNCs with targeted properties. We also address current challenges in translating these design principles into synthetic strategies that reliably produce the intended structures.

2.1. Templated core design

Stability is one of the fundamental considerations in the design of MNCs. From the valence electronic structure point of view, most of the Au nanoclusters are found with even number of valence electrons. The valence electron count can be calculated by $N^* = n - m - q$ (where n, m, and q share the same meaning with those in $[M_n(L)_m]^q$). These valence electrons can be considered to be contributed by stable basic building blocks, including Au₄ tetrahedra,⁵⁷ Au₃ triangles⁵⁸ and Au₁₃ icosahedra,23 which constitute the core structures of Au nanoclusters. Both Au₄ tetrahedra and Au₄ triangles contribute 2 valence electrons (2e⁻) while the Au₁₃ icosahedron contributes 8 valence electrons (8e⁻). These building blocks can fuse and stack with each other, resulting in numerous evolution pathways. These pathways function as templates that guide the prediction, design and synthesis of MNCs with desired sizes and structures. Subsequent composition adjustments of these cores allow further property tailoring for better PL or catalytic performance.

2.1.1. Au₄ and Au₃ as basic building blocks. Au₄ and Au₃ building blocks play vital roles in constructing the MNC core.

Xu et al. observed that all 71 liganded Au nanoclusters reported till then can be decomposed into these two elementary units. ⁵⁹ Typically, Au nanocluster evolution involves the sequential addition of Au_4 or Au_3 units to the existing core backbone, increasing the valence electron number by two. The identification of elementary blocks is evidenced by Au–Au bond length differences. ^{60,61} Many possible spatial arrangement patterns of their stacking and packing have been observed. Fusion of 2 Au_4 units by sharing a common vertex atom to form a bitetrahedral Au_7 is very common, like the Au_7 core of alkynyl-protected Au_{22} ($^tBuPhC \equiv C$)₁₈ ($^tBuPhC \equiv C$ represents 4-(tert-butyl)phenylacetylene). ⁶² These possible arrangements result in numerous evolution pathways of MNCs.

A representative face-centred cubic (fcc) evolution pathway begins with $Au_{20}(SR)_{16}$, 63 progressing through $Au_{21}(SR)_{15}$ to $[Au_{23}(SR)_{16}]^-$, 64,65 where the core expands via symmetric Au_3 additions. 60 As illustrated in Fig. 2a, the Au₁₀ core in Au₂₁(SR)₁₅ and subsequently the Au₁₃ core in [Au₂₃(SR)₁₆] arise from successive additions to a central Au₇ bitetrahedral backbone. A similar template exists featuring double Au₇ in the backbone which underlies $Au_{28}(SR)_{20}$ (8e⁻)⁶⁶ and $Au_{30}(SR)_{18}$ (12e⁻)⁶⁷ (Fig. 2b). Based on these priorly obtained structures, Xiong et al. predicted that the 10e species Au₂₉(SR)₁₉, which was previously detected in mass spectroscopy (MS) experiments by Dass and colleagues, 68 would exhibit an intermediate core structure comprising 2 Au₇ and 1 additional Au₃.⁶⁹ Li et al. later confirmed the hypothesis via SCXRD, which exemplifies the predictive power of templated core design.⁵⁸ Consistently, the HOMO-LUMO gap narrows as the core size of the species in the template increases (~1.7 eV in Au₂₈(SR)₂₀ to 1.25 eV in Au₃₀(SR)₁₈).^{58,66} This evolution mode is further supplemented by the discovery of Au₃₆(SR)₂₂, which features an additional Au₃ unit.70 It would be delightful to see the discovery of more species that behave according to this evolution template.

An alternative uniform anisotropic growth was discovered starting from Au₂₈(TBBT)₂₀ (TBBT stands for 4-tert-butylbenzenethiolate),⁷¹ which grows in 4e⁻ steps through the addition of two Au₄ units per stage. Unlike the parallel Au₇ units in the earlier Au₂₈(SR)₂₀ isomer protected by S-c-C₆H₁₁ (cyclohexanethiolate), this isomer features crossed Au₇ units forming a distinct architecture. Following the successful synthesis of Au₂₈(TBBT)₂₀, Jin and colleagues extended this template to Au₃₆(TBBT)₂₄, ²⁸ Au₄₄(TBBT)₂₈, ⁵⁷ Au₅₂(TBBT)₃₂, ⁶¹ revealing a uniform core expansion via stacking of 2 Au4 units into a double-helix superstructure (Fig. 2c). Optical gaps decrease progressively from Au_{28} , Au_{36} , Au_{44} , to Au_{52} ($E_g = 1.77$, 1.76, 1.51, and 1.39 eV, respectively) as reflected in red-shifted UV-Vis absorption peaks (absorption peaks at 702, 704, 820, and 890 nm, respectively) (Fig. 2e). Notably, all species in the template exhibit chirality with quasi- D_2 symmetry except Au₃₆, whose cubic geometry bestows it higher symmetry with quasi- D_{2d} Au-S framework with no chirality. This template inspired theoretical design of new analogues, such as $Au_{60}(SR)_{36}$, $Au_{68}(SR)_{40}$, and $Au_{84}(SR)_{48}$. 72,73 $Au_{76}(SR)_{44}$ synthesized by Takano et al. 74 was proposed to follow the same trend, exhibiting 9 Au₄ layers.⁷⁵ This evolution template also inspired Liu

 $et\ al.$, who demonstrated rational synthesis of a ${\rm Au_{36}(SR)_{24}}$ isomer based on $de\ novo$ design. The core structure features a 2-dimensional growth pattern proposed based on theoretical calculations. It would be valuable to observe further studies that provide verification for this hypothesized template.

Other less-explored pathways include ring-like tetrahedral growth around a Au_7 backbone, observed in species like $Au_{22}(^tBuPhC \equiv C)_{18}$, $Au_{34}(S-c-C_6H_{11})_{22}$, 77 and $Au_{40}(o-MBT)_{24}$ (o-MBT represents 2-methylbenzenethiolate). However, a structural gap remains between Au_{22} and Au_{34} —potentially bridged by a nanocluster comprising a Au_7 backbone encircled by an incomplete ring of two vertex-sharing tetrahedra—suggesting the possible existence of a novel structural isomer within the $Au_{28}(SR)_{20}$ family.

2.1.2. Au₁₃ **as a building block.** In addition to $2e^-$ building blocks, icosahedral Au₁₃ units ($8e^-$) are vital to the core architecture of many Au nanoclusters, including the flagship nanocluster $[Au_{25}(SR)_{18}]^{-.2}$ Its superior stability can be explained by the superatom model, attributing the closed $8e^-$ shell to a noble-gas-like electronic configuration. Au₁₃ units are observed to connect *via* vertex-sharing (sharing 1 vertex atom) or face-sharing (sharing 3 atoms on a face of the icosahedron) modes to form larger structures.

A vertex-sharing template is evident in mono-, bi-, and triicosahedral species such as $[Au_{13}(dppe)_5Cl_2]^{3+}$, 82 $[Au_{25}(PPh_3)_{10}]$ $(SR)_5Cl_2^{2+}$, and $[Au_{37}(PPh_3)_{10}(SC_2H_4Ph)_{10}Cl_2]^+$ (where dppe = 1,2-bis(diphenylphosphino)ethane, PPh₃ = triphenylphosphine, and SC₂H₄Ph = phenylethylthiolate),²³ respectively (Fig. 2d). Interactions among Au₁₃ units lead to distinct electronic transitions while preserving the electronic features arising from a single Au₁₃ unit. This is exemplified by the UV-Vis absorption peaks at 1230 and 795 nm resulting from the interacting triicosahedron in Au₃₇ and the 670 nm peak from bi-icosahedral interaction in Au₂₅ (Fig. 2f and g). As expected, the optical energy gap decreases with increasing core size ($E_{\sigma} \sim 1.96$ eV, 1.73 eV, and 0.83 eV respectively). As the core evolves further, a pentameric Au₆₀ ring is formed.⁸³ However, variations in the Au₁₃ unit arrangement significantly alter the electronic transition behaviour. The absorption peak arising from the interaction within the pentamer structure is observed at 850 nm, different from the double-peak behaviour in Au₃₇. This suggests that further studies—especially on tetra-icosahedral species-could clarify the evolution of optical properties. In thiolate-protected Au nanoclusters, while mono-icosahedral [Au₂₅(SR)₁₈] and bi-icosahedral Au₃₈SR₂₄ have been discovered, 2,81 larger analogues have not yet been reported. Given the distinct chiral behaviours of [Au_{2.5}(SR)₁₈] and Au₃₈(SR)₂₄, uncovering structures of larger species in the template could yield new insights into chirality trends.

In addition to the abovementioned templates, additional pathways have been proposed but await verification from successful synthesis. For example, a bi-tetrahedral growth template is inferred from the evolution from $\mathrm{Au}_{20}(\mathrm{SR})_{16}$ to the $\mathrm{Au}_{28}(\mathrm{SR})_{20}$ isomer in the first template. Moreover, Au_{11} units are recurring in species like $\mathrm{Au}_{11}(\mathrm{PPh}_3)_7\mathrm{Cl}_3$ and $[\mathrm{Au}_{20}(\mathrm{PPhpy}_2)_{10}\mathrm{Cl}_4]^{2^+}$ (PPhpy₂ = bis(2-pyridyl)-phenylphosphine) where 2 edge-sharing Au_{11} units are observed.^{84,85}

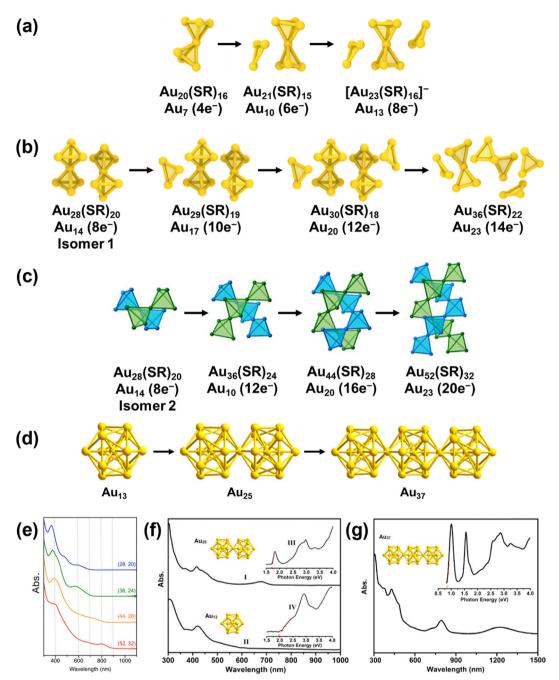


Fig. 2 Illustrations of various core growth templates of Au nanoclusters depicting the Au core structural evolution with stepwise addition of (a) the Au₃ unit onto the Au_7 backbone and (b) the Au_3 unit onto the Au_{14} backbone. (c) Template with stepwise addition of 2 Au_4 units forming a double-helix superstructure, and (e) their respective UV-Vis spectra. The mass formula with Au core size and valence electron number N^* is indicated below each structure. Reprinted with permission from ref. 57. Copyright 2016, American Chemical Society. (d) Template with stepwise addition of Au₁₃, and (f) UV-Vis-NIR spectra of $Au_{25}(I)$ and $Au_{13}(II)$ nanoclusters. Insets: Spectra on the photon energy scale: $Au_{25}(III)$ and $Au_{13}(IV)$. The Au core sizes are indicated below each structure. (g) UV-Vis-NIR spectra of the Au₃₇ nanocluster. Inset: Spectrum on the photon energy scale. Reprinted with permission from ref. 23. Copyright 2015, American Chemical Society.

2.1.3. Core composition modification for property adjustment. Heteroatom doping of the metallic core allows finetuning of MNC properties such as PL and catalytic performance via electronic structure modulation.86,87 According to the twostep spherical jellium model where background potentials attributed to host (e.g. Au) and dopant atoms are different but uniform, 88 when a foreign atom of lower valency is doped into the icosahedron Au₁₃ core, the lower frontier orbitals (1P orbitals) are destabilized and upshifted, while a dopant of higher valency would induce a downshift (Fig. 3a). Elements of larger period number in the same group or smaller group number in the same period tend to expand the HOMO-LUMO

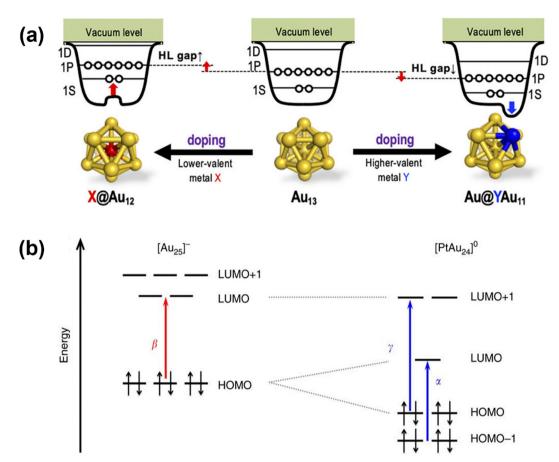


Fig. 3 (a) Illustrations of two-step spherical jellium potentials for dopant Au₁₃ doped with an atom of lower (left) or higher (right) valent element. Reprinted with permission from ref. 88. Copyright 2021, American Chemical Society. (b) Illustrations of the electronic energy levels of undoped [Au₂₅] and doped [PtAu₂₄] 0 . α , β and γ denote different optical transitions which can be observed in their UV-Vis spectra. Reprinted with permission from ref. 87. Copyright 2017, Springer Nature Limited.

gap, which reduces nonradiative decay rates and improves PL quantum yield (QY), consistent with the energy gap law.⁵⁶ As demonstrated by Hirai et al., doping-induced HOMO-LUMO gap modulation leads to predictable monotonic PL peak shifts and QY variation. 89 Doping can also suppress core vibrations in terms of electron-acoustic phonon interactions as shown by Hg- or Cd-doped Au₂₅, thereby enhancing PL intensity when they are incorporated in thin films.⁹⁰

Doping induced HOMO-LUMO gap variation also affects catalytic performance. When a single Pt atom is doped at the centre of the Au₁₃ icosahedral core in [Au₂₅(SC₆H₁₃)₁₈]⁻ (SC₆H₁₃ represents 1-hexanethiol), the 8e core is changed to 6e causing a Jahn-Teller-like distortion of the PtAu12 core accompanied by 1P orbital splitting (Fig. 3b).87 The resultant reduction potential matches well with the reduction potential of a proton, significantly enhancing its electrocatalytic performance in hydrogen evolution reaction. Moreover, due to the difference in electronegativity, the dopant could change the electron density of the MNC, thereby altering the adsorption behaviour during catalysis. Ag₂₅ centrally doped by Au in the Ag₁₃ icosahedral core exhibits significant electron donation from Ag to Au which creates more positively charged surface Ag facilitating the adsorption of electron-rich alkynes in the carboxylation of CO2 with terminal alkynes.91

2.2. Functionalization via ligand shell design

Ligand shell design plays a pivotal role in MNC functionalization for various applications including PL,92 catalysis,93 and biomedicine.13 A wide range of ligand types—thiolates,2 phosphines,⁸⁴ α-alkynyls,⁶² sulfido,⁹⁴ halide,⁸⁰ N-heterocyclic carbene (NHC),95 multiple N donor ligand (MND),96 and carboranes⁹⁷—have been employed in MNC diversification. The primary role of the ligand shell in MNCs is to stabilize the metal core, preventing aggregation and further growth. Taking the thiolate-protected Au nanocluster as an example, the strong Au-SR covalent bonds confer greater stability than the weaker Au-ligand interactions present in Au nanoparticles.98 Using ligands with distinct Au-ligand bonds99 or various ligand-ligand interactions 100,101 can effectively alter MNC stability and enable the formation of different stable sizes. 102 While comprehensive reviews on ligand design strategies exist, 20,27 this section focuses on three key considerations on target nanocluster design for desired properties

integrating recent developments: synergy with the metal core, interactions, and interactions with external intra-shell molecules.

2.2.1. Synergy with the metal core. Firstly, the choice of ligand contributes to the electronic structure of MNCs. This is reflected by the slight redshift of the primary absorption peak of $[Au_{25}(SR)_{18}]^-$ (at ~680 nm) when changing the ligand body from non-aromatic to aromatic ones, and the more obvious redshift after replacing thiolate with selenolate (Fig. 4a).¹⁷ A drastic change is observed when using an α-alkynyl ligand, as shown between Au₄₄(PhC = C)₂₈ (PhC = C represents ethynylbenzene) and Au₄₄(TBBT)₂₈ (Fig. 4b), ¹⁰³ due to the more active participation of the PhC\equiv C ligand in the frontier orbitals endowed by its different binding mode, according to theoretical studies. 104 Furthermore, ligand's contribution to the electronic structure gives rise to ligand-to-metal charge transfer (LMCT) through the Au-ligand bond, which influences MNC PL performance. In [Au₂₅(SR)₁₈]⁻, replacing 1-hexanethiol with more electron-donating ligands such as 1-dodecanethiol and 2-phenylethanethiol (PET) increases fluorescence QY by 2.5 and 5 times respectively. 92 Besides, a delocalized electron from the electron-rich ligand body of glutathione (GSH) can be

directly donated to the metal core of [Au₂₅(SR)₁₈]⁻, boosting QY by ~ 100 times relative to 1-hexanethiol.

Secondly, the choice of ligand determines the geometric structure of the metal core. For example, Au₂₄(SCH₂Ph-^tBu)₂₀ possesses a bi-tetrahedral Au8 core with anti-prismatic facejoint Au₄ units, ¹⁰⁵ whereas Au₂₄(SeC₆H₅)₂₀ shares the same core size but different geometry. Its 2 cross-joint tetrahedral Au₄ units have the same orientation in space (Fig. 4c). 106 The same effects are observed in Au₂₁ nanoclusters where the Au₁₀ core structure differs between S-Adm (adamantanethiolate) and StBu (tert-butylthiol)-protected nanoclusters. 107

2.2.2. Interactions within the ligand shell. Interactions within the ligand shell critically shape the overall structure and properties of MNCs. Rigid ligands such as alkynyls introduce a strong steric hindrance effect that induces distinct structural conformations in the ligand shell. This is exemplified by $[Au_{2.5}(C \equiv CAr)_{1.8}]^-$ (Ar = 3,5-bis(trifluoromethyl)phenyl) whose 3 of the 12 Au atoms in staple motifs are twisted about 60° compared to its thiolate-protected counterpart, giving rise to its intrinsic chirality with a D_3 symmetry (Fig. 5a and b). ¹⁰⁸ Ligand rigidity can also influence the thermodynamically stable surface structures of MNCs. In the case of Au₂₈(SR)₂₀, S-c-C₆H₁₁

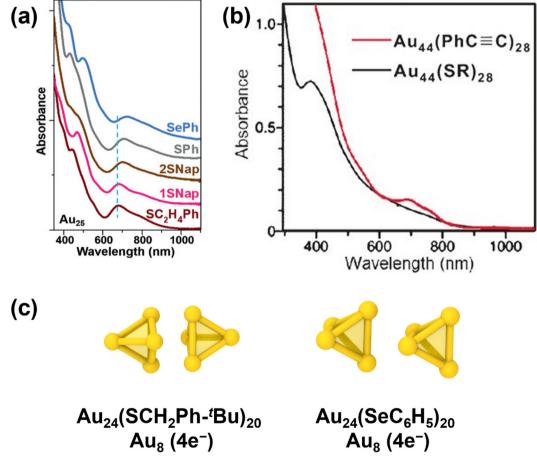


Fig. 4 (a) UV-Vis absorption spectra of Au₂₅ nanoclusters protected by various thiolates or selenolate. Reprinted with permission from ref. 17. Copyright 2021, Wiley-VCH. (b) UV-Vis spectra of Au₄₄ nanoclusters in CH₂Cl₂. Reprinted with permission from ref. 103. Copyright 2017, Wiley-VCH. (c) Illustration of the core structures of Au₂₄ nanoclusters. The mass formula with Au core size and valence electron number N* is indicated below each structure.

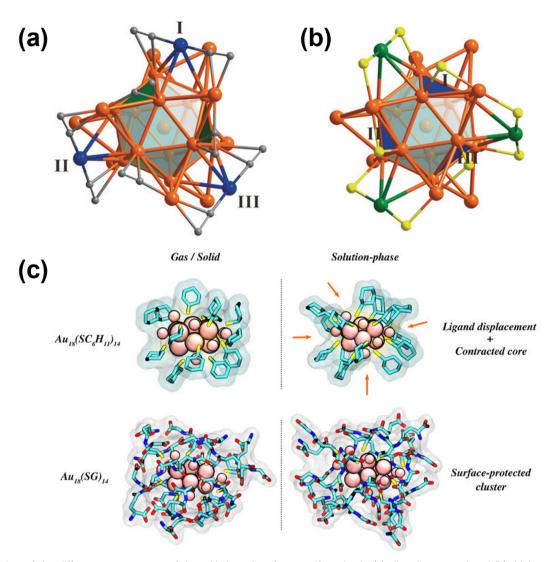


Fig. 5 Illustration of the different arrangements of three V-shaped surface motif staples in (a) alkynyl-protected and (b) thiolate-protected Au₂₅ nanoclusters. Colour label: orange, blue and green spheres, Au; yellow spheres, sulfur; grey spheres, carbon. (c). Reprinted with permission from ref. 108. Copyright 2018, Wiley-VCH. Simulated structures of S-c-C₆H₁₁-protected and SG-protected Au₁₈(SR)₁₄ in the gas or solution phase. Reprinted with permission from ref. 100. Copyright 2018, American Chemical Society.

stabilizes more protruded staple motifs compared to TBBT, which enhances carbon monoxide (CO) adsorption at the motif Au atoms during catalytic CO oxidation reaction.⁶⁶

Ligand-ligand interactions can also be utilized to rigidify the ligand shell for PL enhancement effects. Stronger π - π stacking interactions between the adjacent aromatic ligands, in addition to the bulkier ligand body, contribute to the more rigidified structure which suppresses the high-frequency optical phonons in [Au₂₅(SR)₁₈]⁻ protected by aromatic ligands, leading to enhanced PL compared to [Au₂₅(SR)₁₈] protected by non-aromatic ligands. 109 Likewise, the rigid surface and strong internal π - π stacking interactions in NHC-stabilized Au₁₃ result in superior QYs of 16%, much higher than those of other nanoclusters with the same Au₁₃ core. 110 Ligandligand hydrogen bonds can also provide surface structural rigidity. The extensive intermolecular hydrogen bonds between the H-donors (-OH and -NH) and the H-acceptor (C=O) functional groups between adjacent ligands led to smaller structural changes between the solid and solution phases for glutathione-protected Au₁₈(SR)₁₄ as compared to the S-c-C₆H₁₁-protected nanocluster (Fig. 5c), and superior PL performance. 100

2.2.3. Interactions with external molecules. The third key consideration in ligand design is how the ligand shell interacts with external molecules. These interactions—occurring through both physical and chemical means—critically influence MNC properties. 111,112 Strategic engineering of these interactions allows researchers to expand MNC functional capabilities beyond those achievable through the metal core and intra-shell design alone.

MNC properties can be modulated through carefully designed electrostatic interactions. For instance, ligand's electrostatic interactions with cationic surfactant CTA+ (cetyltrimethylammonium) induced [Au₂₅(p-MBA)₁₈]⁻ (p-MBA stands

for para-mercaptobenzoic acid) isomerization. 113 By means of the synergistic effect of electrostatic interactions (between the deprotonated carboxylic groups and the positively charged ammonium headgroup) and $CH \cdots \pi$ interactions (between the aromatic ring of p-MBA ligands and the small carbon tails at the ammonium headgroup), the CTA⁺ ions adsorbed onto the nanocluster surface, forming a double layer structure (Fig. 6a). The resultant surface rigidification effect stretches and rotates the inner metal core, forming a new isomer with distinct optical properties (Fig. 6b). Likewise, multi-layer ligand engineering was applied to ATT (6-Aza-2-thiothymine)-stabilized Au₁₀ through hydrogen bonding with ARG (L-arginine), which is further ion-paired with TOA⁺ (tetraoctylammonium), effectively suppressing kernel vibrations. 114

Intercluster interactions are pivotal to MNC self-assembly behaviour and supercrystal formation. [Au₂₅(p-MBA)₁₈] crystallization was facilitated by combined $CH \cdots \pi$ and ionparing interactions between p-MBA ligands at the apex position and tetraethylammonium cations (TEA⁺) (Fig. 6c). 116 These interactions achieved an intricate balance, detaching the SR- $[Au(I)-SR]_2$ motifs from the surface of $[Au_{25}(p-MBA)_{18}]^-$ to form SR-[Au(I)-SR]₄ linkers that connected adjacent distorted monomers into an orderly three-dimensional architecture. Similarly, controlled van der Waals interactions between the surface ligands of Au₂₉(S-Adm)₁₉ enabled helical assembly of the MNC monomers.⁵⁸ These ordered superstructures serve dual functions: facilitating structural determination for structure-property relationship construction and inducing synergistic properties distinct from those of individual MNC monomers. The dual ligand system in Au₄Ag₁₃(dppm)₃(SR)₉ (dppm = bis(diphenylphosphino)methane and SR specifically denotes 2,5-dimethylbenzenethiolate in this case) exemplifies this principle, where six pairs of intercluster $CH \cdot \cdot \cdot \pi$ interactions between dppm aromatic rings and SR aromatic hydrogens promoted supercrystal formation while compact packing significantly enhanced radiative transitions through intramolecular vibration/rotation restriction.117 Interestingly, the supercrystal formed by $CH \cdots \pi$ and $\pi \cdots \pi$ interactions between dppp (1,3-bis-diphenylphosphine propane) ligands in [Pt₁Ag₁₈-(S-Adm)₂(dppp)₆Cl₆]²⁺ not only exhibited crystallization-induced PL enhancement but also demonstrated promising optical waveguide performance with low optical loss and polarized emission, attributed to its distinct packing mode. 118

Another crucial consideration is ligands' interaction with the solvent environment, especially for bio-related applications. Ligands such as peptides, DNA, and proteins contribute to biocompatibility necessary in applications such as biosensing, bioimaging and photothermal therapy. 119-122 For inherently hydrophobic ligands such as NHC, biocompatibility can be achieved through functionalization with polar water-soluble functional groups such as triethylene glycol monomethyl ether. 123 Bioconjugation represents another effective strategy to achieve biocompatibility. Zhang et al. demonstrated stoichiometric conjugation between [Au₂₅(SR)₁₈]⁻ and BSA (bovine serum albumin) via electrostatic interactions and hydrogen bonds which simultaneously enhanced NIR-II emission and potential theranostic applicability. 124 Host-guest chemistry offers additional opportunities, as demonstrated by complex formation between β-cyclodextrin (CD) and the 4-(tert-butyl)benzyl mercaptan protected Au₂₅ nanocluster. 125

Exploiting the chemical reactivity of ligands also serves as a powerful tool for property enhancement and functional diversification. Deng et al. demonstrated that intracluster crosslinking between GSH ligands in Au₂₂(SG)₁₈ via Bis-Schiff base linkage formation enhanced the PL QY over 11-fold. 126 Similarly, [Au₁₁(PNHP)₄Br₂]⁺ (PNHP stands for [PPh₂(CH₂)₂]₂NH where PPh2 represents diphenylphosphine) was functionalized through amidation with acyl chlorides to simultaneously introduce chirality and desired functional groups. 127

Recent advances include click chemistry-compatible MNCs, such as $[Au_{25}(SR)_{18}]^-$ protected by azide-functionalized thiolate (SCH₂CH₂-p-C₆H₄-N₃). 128 Its reactivity in strain-promoted azido-alkyne cycloaddition (SPAAC) was found to be affected by ligand regioisomerism with the para isomer exhibiting the highest reaction rates while the ortho isomer failed to stabilize the nanocluster during reaction. 129 In a complementary approach, reactivity in SPAAC is incorporated into the DNA-Ag₁₆ nanocluster by attaching ring-strained alkyne bicyclononyne to the DNA ligands (Fig. 6d). 130 When conjugated with azido-modified human insulin, the nanocluster maintained its original photophysical properties while enabling specific staining of the Chinese hamster ovary membranes with promising stability.

Interactions in terms of molecular adsorption are crucial in designing MNCs with desired catalytic performance. Liu et al. reported enhanced OH⁻ adsorption during oxygen evolution reaction in alkaline medium by ligands with stronger electronwithdrawing capability. 131 Compared to [Au₂₅(SR)₁₈] protected by MHA (6-mercaptohexanoic acid) and H-cys (homocysteine) (Fig. 6e), p-MBA in $[Au_{25}(p-MBA)_{18}]^-$ induced more positively charged Au(1) active sites in the surface motifs (Fig. 6f), rendering nearly 4 times catalytic performance enhancement (Fig. 6g). Through Tafel slope analysis and in situ Raman spectroscopy, the authors determined that the difference in charge density around the active site altered the rate determining step: decomposition of Au–O–OH for $[Au_{25}(p-MBA)_{18}]^-$ versus deprotonation of Au-OH for the other nanoclusters.

Route development

Route development, the second critical step in synthesis planning, involves designing the reaction pathway from commercially available starting materials to the desired target material. In organic chemistry, this step typically employs retrosynthetic analysis-working backward from the target molecule to identify viable precursors based on established reaction libraries. Multiple possible routes are then evaluated using criteria such as cost-effectiveness, safety, and synthetic feasibility. 41,132 The synthesis of ketoprofen, a nonsteroidal anti-inflammatory drug, illustrates this approach. To avoid regulated substances, a viable retrosynthetic pathway

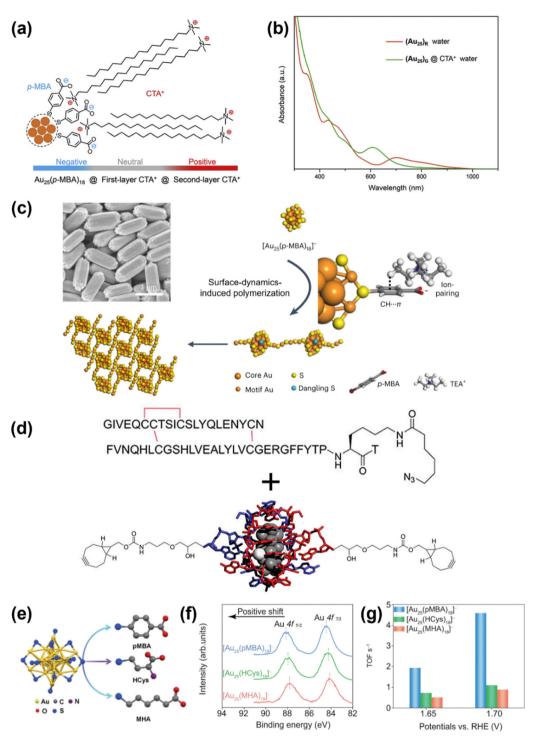


Fig. 6 (a) Schematic illustration of the adsorption of CTA⁺ ions onto the $[Au_{25}(p-MBA)_{18}]^-$ nanocluster surface. (b) UV-Vis absorption spectra of $[Au_{25}(p-MBA)_{18}]^-$ MBA)₁₈]⁻ and [Au₂₅(p-MBA)₁₈]⁻ after isomerization. Reprinted with permission from ref. 113. Copyright 2021, Elsevier. (c) Field-emission scanning electron microscopy (FESEM) image of hexagonal rod-like supercrystals formed by $[Au_{25}(p-MBA)_{18}]^-$ and the schematic illustration of the surface rearrangement and crystallization processes induced by $CH\cdots\pi$ and ion-paring interactions between the ligand at the apex position and TEA^+ . Reprinted with permission from ref. 116. Copyright 2023, Springer Nature Limited. (d) Schematic illustration of the conjugation between human insulin and the bicyclononyneterminated DNA-Ag₁₆ nanocluster. Reprinted with permission from ref. 130. Copyright 2023, American Chemical Society. (e) Schematic illustration of [Au₂₅(SR)₁₈] annoclusters protected by various ligands, (f) their high-resolution Au 4f XPS (X-ray photoelectron spectroscopy) spectra, and (g) their turnover frequency values at 1.65 and 1.7 V. Reprinted with permission from ref. 131. Copyright 2023, Springer Nature Limited.

begins with commercially available $\alpha,\beta\text{-dibromopropanoate}$ and benzophenone. These precursors initially undergo

dehydrohalogenation and iodination, respectively, followed by a coupling reaction. The synthesized intermediate is

subsequently hydrolysed and hydrogenated to yield the desired pharmaceutical product. 133

MNC synthesis planning adopts a similar conceptual framework but requires specialized approaches due to fundamental differences in reaction mechanisms. Unlike organic synthesis with its discrete covalent bond transformations, MNC formation involves complex formation processes with numerous concurrent reactions and intermediates. While organic chemistry can draw upon extensive libraries of well-characterized reactions, the high complexity and rapid kinetics of nanocluster formation have hindered the development of analogous elementary reaction libraries for MNCs.

Despite these challenges, researchers have developed qualitative "reaction maps" that guide MNC synthesis by correlating starting materials, intermediates, and conditions with structural outcomes. A typical MNC synthesis protocol adapted from the Brust-Schiffrin method involves metal-ligand complex formation through ligand exchange with metal salts (e.g., HAuCl₄ with thiols), followed by reduction (often with NaBH₄) to generate core-shell nanoclusters. 134

Several distinct synthetic strategies have emerged for controlling specific MNC structural features. Direct synthesis through one-step reduction produces numerous welldefined nanoclusters like $[Au_{25}(SR)_{18}]^-$ and $[Ag_{44}(SR)_{30}]^{4-}$ through reduction-growth processes. 29,135,136 "Size-focusing" approaches convert polydisperse mixtures into monodisperse products by leveraging thermodynamic stability differences. 137 Seeded growth reactions use existing monodisperse MNCs as templates for larger structures, while oxidative etching provides routes to reduce core size. 138 Surface-induced transformations offer pathways to core structure modification through physical means or chemical approaches like ligand exchange. 113,139 Core composition can be manipulated via co-reduction, ¹⁵⁸ metal exchange with preformed MNCs, 140 or intercluster reactions, 141 each offering specific advantages for dopant

In this section, we systematically examine these synthetic routes for controlling four critical aspects of MNC structure: core size, core structure, core composition, and ligand shell structure. For each approach, we discuss the mechanistic principles and the criteria for selecting optimal routes based on target design requirements.

3.1. Synthesizing the desired core size

MNC synthesis typically employs the reduction of metal-ligand (M-L) complex precursors following adaptations of the Brust-Schiffrin method. During this process, a rich library of species is formed whose N^* s are found to follow an evolution path of $0e^- \rightarrow 2e^- \rightarrow 4e^- \rightarrow 6e^- \rightarrow 8e^- \rightarrow 10e^- via$ a stepwise $2e^$ hopping mechanism based on electrospray ionization mass spectrometry (ESI-MS) and UV-Vis absorption spectroscopy results.^{29,142} While reduction-growth reactions promote formation of larger species from smaller ones, concurrent reactions like isoelectronic addition, disproportionation, and comproportionation are also taking place (Fig. 7a and b). These reactions drive the formation of the thermodynamically

favourable product in a "size-focusing" manner, 137 as exemplified by the synthesis of $[Au_{25}(m-MBA)_{18}]^-$ (m-MBA represents meta-mercaptobenzoic acid) (Fig. 7c).²⁹ Due to the complexity of the reduction process, reaction intermediates can be stabilized via kinetic control methods by carefully adjusting reaction condition parameters (detailed in Section 4). Following synthesis, various separation and purification techniquesincluding solvent fractionation, 143 solvent extraction, 144 thinlayer chromatography (TLC), 145 high-performance liquid chromatography (HPLC), 146 and electrophoresis 135—can be applied to remove impurities and obtain product monodispersity. This one-step direct reduction approach has successfully yielded MNCs of varied sizes, compositions, and ligands, such as thiolate-stabilized $\text{Au}_{15}(\text{SR})_{13}$, 33 $\text{Au}_{18}(\text{SR})_{14}$, 33 $\text{Au}_{20}(\text{SR})_{16}$, 35 $[\text{Au}_{27}(\text{SR})_{13}]^{4+}$, 35 $[\text{Au}_{25}(\text{SR})_{18}]^{-29,135}$ $\text{Au}_{36}(\text{SR})_{22}$, 70 $[\text{Ag}_{44}(\text{SR})_{30}]^{4-}$, 136 DNA-stabilized Ag_{16} , 147 Ag_{11} , 148 and amidinate-protected Cu_{11} , 149 just to name a few.

As an alternative to direct synthesis, a two-step "sizefocusing" route can be employed. 137 This approach begins with generating a mixture of species through reduction-growth, followed by an additional step that adjusts reaction kinetics to favour more thermodynamically stable products. 150 For example, introduction of an excess ligand can accelerate the etching reaction and promote the "size-focusing" process. Sizemixed Au_x(TBBT)_y obtained from NaBH₄ reduction of Au-TBBT complexes can be reacted with excess TBBT thiol at elevated temperature to form Au₅₂(TBBT)₃₂.61 Similar approaches have enabled the synthesis of $Au_{36}(DMBT)_{24}$ (DMBT = 3,5dimethylbenzenethiol).76 Alternatively, introducing different ligands can initiate simultaneous etching and ligand exchange, as demonstrated by the formation of Au₃₈(SC₁₂H₂₅)₂₄¹⁵¹ and $Au_{21}(S^tBu)_{15}$.60

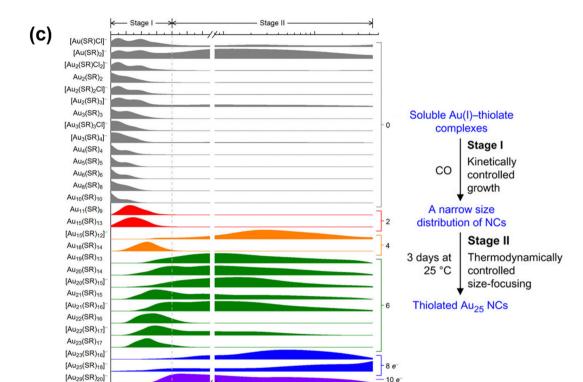
Monodisperse MNCs can serve as seeds for continued reduction-growth reactions to yield larger core sizes. Yao et al. successfully synthesized Au₃₈(SR)₂₄ and [Au₄₄(SR)₂₆]²⁻ by adding [Au₂₅(SR)₁₈]⁻ as seed to the Au(1)-SR complex precursor followed by CO-mediated reduction. 138 [Au25(SR)18] played a duel role in this process—it reacts with the Au(1)–SR complexes or nanocluster species while also adsorbing CO molecules, making CO more susceptible to oxidation as evidenced by the successful detection of [Au₂₅(SR)₁₈CO]⁻ via ESI-MS. Monitoring reaction intermediates revealed two parallel size growth patterns: LaMer-like monotonic size growth and volcano-shaped aggregative growth, both following the 2e hopping mechanism (Fig. 7d). Fine-tuning of the reaction kinetics allowed optimization toward Au₃₈(SR)₂₄ as the major product. This seeded growth approach has also been successfully applied to silver nanoclusters, producing Ag₅₀(dppm)₆(SR)₃₀ $[Ag_{44}(SR)_{30}]^{4-.152}$

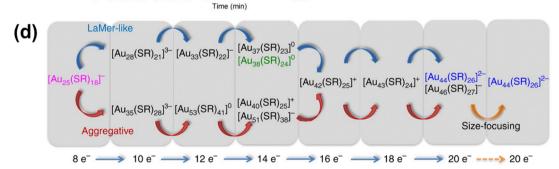
Conversely, oxidative etching provides an effective method for reducing MNC core size, generating MNCs with higher ligand-to-metal ratios. In thiol-mediated etching of larger Au nanoparticles, oxygen molecules initiate the process by radicalizing thiol molecules to form thiyl and peroxy radicals. These radicals cleave surface motifs and oxidize Au(0) to form surfaceexposed Au(1), creating new Au(1)-S bonds (Fig. 8a). 153 A similar

(a) Reduction:
$$Au_6(SR)_6 + CO + 4 HO^- \longrightarrow Au_6(SR)_4 + 2 - SR + CO_3^2 + 2 H_2O$$

Growth: $Au_6(SR)_4 + Au_5(SR)_5 \longrightarrow Au_{11}(SR)_9$

Overall:
$$Au_6(SR)_6 + Au_5(SR)_5 + CO + 4 HO^- \longrightarrow Au_{11}(SR)_9 + 2 -SR + CO_3^{2-} + 2 H_2O$$





1000

Fig. 7 Proposed reaction schemes of (a) reduction-growth formation of Au₁₁(SR)₉ and (b) possible isoelectronic addition, disproportionation, and comproportionation reactions that occur during the size-focusing stage. (c) ESI-MS spectral profiles with normalization of the intensity of the complex precursors and nanocluster species throughout the synthesis (left) and the overall reaction scheme (right). Reprinted with permission from ref. 29. Copyright 2014, American Chemical Society. (d) Schematic illustration of the growth pathways from $[Au_{25}(SR)_{18}]^-$ to $[Au_{44}(SR)_{26}]^{2-}$ featuring two growth patterns. Reprinted with permission from ref. 138. Copyright 2017, Springer Nature Limited.

etching of [Au₂₅(SR)₁₈] using excessive thiols. He had mechanistic reaction stages: decomposition and recombination (Fig. 8b).

10 20 30

40 50 60

100

radical-induced mechanism takes place during the oxidative investigations reveal that the reaction proceeds through two

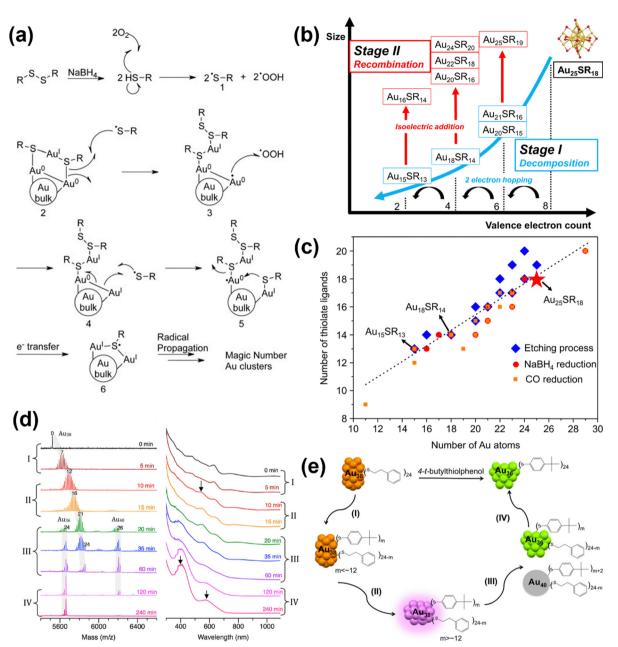


Fig. 8 (a) Proposed mechanism of Au nanoparticle etching by thiol. Reprinted with permission from ref. 153. Copyright 2015, Wiley-VCH. (b) Schematic illustration of the etching process of [Au₂₅(SR)₁₈] which depicts two reaction stages: decomposition and recombination. (c) Comparison of the distinct nanocluster species formed during etching and reduction-growth processes of $[Au_{25}(SR)_{18}]^{-}$. The black dashed line is a guideline for reference. Reprinted with permission from ref. 154. Copyright 2021, Springer Nature Limited. (d) ESI-MS spectra (left) and the corresponding UV-Vis spectra (right) characterized at various points of time during the transformation reaction. Three grey shadows highlight three groups of peaks: $Au_{36}(TBBT)_m(PET)_{24-m}$. $Au_{38}(\text{TBBT})_m(\text{PET})_{24-m}, \text{ and } Au_{40}(\text{TBBT})_m(\text{PET})_{24-m}, \text{ from left to right, respectively. The number of TBBT ligands exchanged onto the cluster (\textit{m}) is the following the following properties of the contract of the cluster (\textit{m}) is the following properties of the cluster (\textit{m}) is the cluste$ highlighted on top of the mass peaks. (e) Schematic reaction pathway showing conversion from Au₃₈(PET)₂₄ to Au₃₆(TBBT)₂₄ which exhibits four stages: ligand exchange, structure distortion, disproportionation, and size-focusing. Reprinted with permission from ref. 139. Copyright 2013, American Chemical Society

The initial decomposition stage follows a reverse 2e hopping mechanism compared to the reduction-growth process. Subsequently, decomposition products recombine with Au(1)-SR complexes to form isoelectronic Au nanoclusters with identical N* values but higher SR: Au ratio—species not observed in simple reduction-growth processes (Fig. 8c). This allows the successful synthesis of $Au_{25}(SR)_{19}$ from $[Au_{25}(SR)_{18}]^{-}$, ¹⁵⁵ which

is predicted to possess a smaller core, consisting of one Au₄ unit alongside a bitetrahedral Au₇. 156

Intercluster reaction offers another approach to modulate MNC core size. When $[Au_{25}(SR)_{18}]^-$ is oxidized to the neutrally charged Au₂₅(SR)₁₈, it loses the stability endowed by the shellclosing 8e⁻ valence electron count. An elevation in temperature then initiates the intercluster fusion reaction forming

Au₃₈(SR)₂₄ from two Au₂₅(SR)₁₈ nanoclusters. ¹⁵⁸ Remarkably, the Au₁₃ icosahedral cores of individual Au₂₅(SR)₁₈ integrate to form Au₃₈(SR)₂₄ with a face-fused Au₂₃ biicosahedral core.

Beyond redox reactions and intercluster reactions, the synergy between surface ligands and metal cores can induce size modification during ligand exchange processes. For example, treating Au₃₈(PET)₂₄ with TBBT in large excess results in complete ligand exchange to produce Au₃₆(TBBT)₂₄. ¹³⁹ Timedependent ESI-MS and UV-Vis characterization revealed four distinct reaction stages in this transformation (Fig. 8d and e):

- (1) Initial ligand exchange forming $Au_{38}(TBBT)_m(PET)_{24-m}$ (m < 12).
- (2) Progressive structural distortion of Au₃₈(TBBT)_m- $(PET)_{24-m}$ likely triggered by steric interactions from the bulky TBBT ligands.
- (3) Disproportionation of structurally distorted Au₃₈*(SR)₂₄ to form $Au_{36}(SR)_{24}$ and $Au_{40}(SR)_{26}$ through internal reconstruction.
- (4) Simultaneous ligand exchange and size-focusing, yielding Au₃₆(TBBT)₂₄ with approximately 90% yield.

This methodology has been extended to synthesize $Au_{21}(S-Adm)_{15}$ and $Au_{16}(S-Adm)_{12}$ from $Au_{18}(S-c-C_6H_{11})_{14}$ and Au₁₅(SG)₁₃, respectively, utilizing the bulky HS-Adm ligand. 107,159 Notably, unlike the Au₃₆(TBBT)₂₄ case, these products exhibit larger core sizes than their predecessors, indicating that the product core size is highly ligand-dependent.

3.2. Tuning for the desired core structure

Once the desired core size is achieved, further refinement of the core structure—with minor core size adjustments that preserve the valence electron count-can be accomplished through various surface-induced transformation methods. These approaches can be broadly categorized into physical measures and chemical approaches.

Firstly, physical measures can effectively modify core structures without changing chemical composition. As discussed previously, pairing anionic surface ligands with bulky cations promotes electrostatic and $CH \cdot \cdot \pi$ interactions that rigidify the nanocluster surface, inducing core rotation in [Au₂₅(SR)₁₈]⁻. ¹¹³ Similarly, modulating solvent pH can regulate inter-ligand hydrogen bonds to trigger isoelectronic transformations. For example, by adjusting the solvent pH from 5.5 to either 2.5 or 8, Au₂₂(SG)₁₈ is readily converted to isoelectronic Au₂₄(SG)₂₀ or Au₁₈(SG)₁₄ respectively. 160 Although the exact structures of these glutathione-protected species have not been fully resolved, their UV-Vis absorption profiles—reliable indicators of their core size and structure-closely resemble those of structurally characterized analogues with different ligands. This suggests that they likely possess similar core structures: bitetrahedral Au₈ in Au₂₄, bitetrahedral Au₇ in Au₂₂, and bioctahedral Au₉ in Au₁₈ (Fig. 9a). 105,161-163

Surface dynamics can be strategically altered through ligand exchange reactions to induce core structure modifications. Reacting $Au_{30}(S^tBu)_{18}$ with excessive HSPhX (X = -H or -^tBu) induces inter-template core structure conversion, forming monodisperse Au₃₆(SPhX)₂₄. ¹⁶⁴ This transformation fundamentally reconfigures the 12e Au₂₀ in Au₃₀(S^tBu)₁₈ (Fig. 2b)—which can be regarded as two Au₃ units attached to two bitetrahedral Au₇ units—into 2 crossing vertex-fused tritetrahedral Au₁₀ in Au₃₆(SPhX)₂₄ (Fig. 2c). Notably, this process is reversible—the original Au₃₀(S^tBu)₁₈ can be regenerated by exchanging the surface ligand back to tert-butylthiol. Similarly, the $8e^- Au_{13}$ core in $[Au_{23}(S-c-C_6H_{11})_{16}]^-$ can transform to the double bitetrahedral Au₁₄ core of Au₂₈(TBBT)₂₀ (Fig. 9b). ¹⁶⁵ The authors propose that progressive replacement of S-c-C₆H₁₁ with TBBT alters surface dynamics, inducing core distortion that ultimately results in core transformation once a threshold concentration of TBBT on the surface is reached.

In addition to ligand exchange, surface motif exchange provides another effective approach for tuning the core structure through surface dynamics modification. Reacting Au(1)-SR complexes with [Au23(SR)16] can transform the cuboctahedral Au_{13} core into an icosahedron, forming $[Au_{25}(SR)_{18}]^{-166}$ Through hetero-ligand experiments, the authors proposed that the transformation process is initiated by the association of 2 SR-[Au(1)-SR]₂ motifs. The process involves disruption of the original longer SR-[Au(1)-SR]₃ motifs and ejection of SR-Au(1)-SR motifs, giving rise to an enhanced degree of freedom in the metal core which facilitates the core conversion.

3.3. Synthesizing the desired core composition

Controlling core composition through heteroatom doping is achievable via three primary routes: one-step co-reduction, twostep metal exchange with monodisperse MNCs, and intercluster reactions. These approaches have successfully incorporated various heteroatoms—including Au, 167 Ag, 168 Cu, 169 Pt, 118 Pd, ¹⁷⁰ Cd, ¹⁷¹ Hg, ¹⁷² Ru, ⁸⁶ Rh, ⁸⁶ Ir, ⁸⁶ and H^{173,174}—into MNCs. Detailed structural analysis revealed that different dopants have distinct preferential sites. In the case of [MAu₂₄(SR)₁₈] (where M represents a dopant), Pt and Pd preferentially replace the central atom in the icosahedral core, Ag, Cd and Hg tend to occupy positions on the icosahedral shell, and Cu typically localizes in the staple motifs. 171,175,176 Such site preference is observed in $Au_{16}Cu_6(^tBuPhC \equiv C)_{18}$, where all 6 Cu atoms are located in the surface motifs as compared to its isostructural counterpart $Au_{22}(^{t}BuPhC \equiv C)_{18}.^{62}$

The one-step co-reduction approach involves mixing precursors of both host and dopant metals—either as metal salts or as metal-ligand complexes—to undergo simultaneous reduction. This facile method enables single-atom doping in various systems, including the Au_{13} icosahedral core of $[Au_{25}(SR)_{18}]^$ with Pt and Pd, 51,177 and phosphine-protected Au₁₃ MNCs with Ru, Rh and Ir.86 If necessary, ligand shell modification can be thereafter conducted. 178 When doping heteroatoms that are homologous to the host metal, such as Ag and Au, multi-doped MNC mixtures often form, such as $[Ag_xAu_{25-x}(SR)_{18}]^-$ and Ag_xAu_{38-x}(SR)₂₄. ^{168,179} HPLC can then be employed to isolate species with different dopant number from the mixture. 180,181 Notably, heteroatom doping can lead to surface reconstruction of the host MNC. Li et al. demonstrated that adding Cd(II)-(SR)₂ to the Au precursors yields $[Au_{23-x}Cd_x(SR)_{16}]^ (x \approx 2)$ which exhibits a structure similar to [Au23(SR)16] but with 2 surface

Review

(a) Au₂₄(SR)₂₀ Au₂₂(C≡CR)₁₈ Au₁₈(SR)₁₄ Au₈ (4e⁻) $Au_{7} (4e^{-})$ Au₉ (4e⁻) (b) 2e' fragment formation Au₁₃ kernel: Edge sharing Au23(SR)16 Au, core bitetrahedron Au₃ triangular unit Au hub atom Twisting of Au, bitetrahedron units

Fig. 9 (a) Illustration of the core structures of $Au_{24}(SR)_{20}$, $Au_{22}(C \equiv CR)_{18}$, and $Au_{18}(SR)_{14}$. The mass formula with Au core size and valence electron number N^* is indicated below each structure. (b) Proposed reaction pathway of conversion of the Au core from $[Au_{23}(S-c-C_6H_{11})_{16}]^-$ to $Au_{28}(TBBT)_{20}$. Reprinted with permission from ref. 165. Copyright 2020, American Chemical Society.

Au atoms replaced by Cd. 171 Increasing the Cd(II)-(SR)2 precursor concentration results in [Au₁₉Cd₂(SR)₁₆]⁻, where each metallic Cd replaces 2 surface Au atoms—preserving the Au₁₃ core while reconstructing the surface motif configurations (Fig. 10a).

Au₁₄ kernel: Au28(SR)20

The second approach employs two-step metal exchange with preformed monodisperse MNCs. After synthesizing $[Au_{24}(PPh_3)_{10}(SC_2H_4Ph)_5Cl_2]^+$, addition of MCl salt (M = Ag/Cu) gives rise to single-atom doping where the dopant occupies the vertex position in the core, displacing one Au atom to the centre (Fig. 10b). 140 Likewise, CdAu₂₄(SR)₁₈ and HgAu₂₄(SR)₁₈ can be obtained by reacting Cd2+ and Hg2+ ions with $[Au_{25}(SR)_{18}]^{-.182}$ It is intriguing that the less noble Hg is reduced by Au in Au25, seemingly contradicting the galvanic sequence. This "anti-galvanic" behaviour is attributed to the decreased oxidation potential of these MNCs falling below the reduction potential of the dopant ions. 183 Metal-ligand complexes can also serve as effective dopant sources. Bakr and colleagues demonstrated that reacting [Ag₂₅(SR)₁₈] with Au complexes replaces the central Ag yielding [Ag24Au(SR)18] and

achieving single-atom doping in Au/Ag bimetallic nanoclusters. 167 Mechanistic studies by Zheng et al. using realtime monitoring in hydrophilic systems suggest that the doped Au atom initially replaces an atom in the surface motif; subsequently, it diffuses dynamically to the icosahedral shell, and ultimately occupies the energetically favourable central position.¹⁸⁴ However, exchange with complexes can also lead to surface motif replacement, as demonstrated when appropriate amounts of Au(1)-SR substitute all the surface motifs of $[Ag_{44}(p\text{-MBA})_{30}]^{4-}$ forming $[Ag_{32}Au_{12}SR_{30}]^{4-}$. 185

Precursor concentration is critical in metal exchange reactions. In the exchange reaction between [Au₂₃(SR)₁₆] and Ag(I)-SR, light doping allows the formation of $[Au_{23-x}Ag_x(SR)_{16}]^{-1}$ (x = 1 to 2) with Ag occupying core vertex positions, which can subsequently form $[Au_{21}(SR)_{12}(P-C-P)_2]^+$ with $Au_2Cl_2(P-C-P)$ (P-C-P stands for bis(diphenylphosphino)methane). Conversely, heavy doping leads to structural conversion, forming $[Au_{25-x}Ag_x(SR)_{18}]^-$ (x ranging from ~4 to ~19). 186,187 In this process, Ag dopants preferentially occupy icosahedral core vertices before filling surface motif positions (Fig. 10c). It

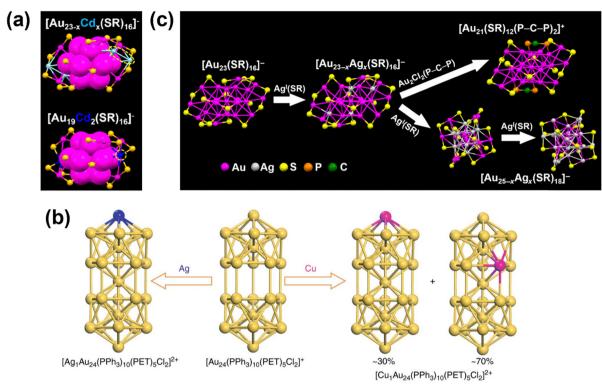


Fig. 10 (a) Structures of [Au_{23-x}Cd_x(SR)₁₆]⁻ and [Au₁₉Cd₂(SR)₁₆]⁻. Colour label: magenta, Au; yellow, S; blue, Cd; light blue, partial occupancy of Cd/Au. Reprinted with permission from ref. 171. Copyright 2017, American Chemical Society. (b) Reaction pathway showing core structures of Au₂₄ and the products of single-atom doping with Ag and Cu. Colour label: yellow, Au; blue, Ag; magenta, Cu. Reprinted with permission from ref. 140. Copyright 2017, Springer Nature Limited. (c) Reaction pathway of $[Au_{23}(SR)_{16}]^-$ with light doping forming $[Au_{23-x}Ag_x(SR)_{16}]^-$ (x = 1 to 2) and thereafter forming $[Au_{21}(SR)_{12}(P-C-P)_2]^+$ and heavy doping forming $[Au_{25-x}Ag_x(SR)_{18}]^-$ upon exchange reaction with Ag(i)-SR. Reprinted with permission from ref. 186. Copyright 2017, The American Association for the Advancement of Science.

should also be noted that the choice of dopant precursor form (ions versus complexes) can impact the synthesis outcome. While CdAu₂₄(SR)₁₈ can be synthesized from both Cd²⁺ ion and complex precursors, 182,188 these precursors yield different products when reacted with [Au₂₃(SR)₁₆]⁻. ¹⁸⁹ Such nanocluster dependency was demonstrated by Zhu et al., who showed that Cd²⁺ ions induce conversion from [Au₂₃(SR)₁₆]⁻ to Au₂₈(SR)₂₀ while Cd(II)-(SR)₂ results in Au₂₀Cd₄(SH)(SR)₂₉.

Composition adjustment can also be realized via intercluster reactions. ¹⁴¹ By reacting $[Au_{25}(SR)_{18}]^-$ with $[Ag_{25}(SR)_{18}]^-$ in different molar ratios, the entire range of alloy composition $[Ag_{25-x}Au_x(SR)_{18}]^-$ (x = 1-24) becomes attainable. 190 Researchers successfully captured the formation of [Ag₂₅Au₂₅(SR)₃₆]²⁻ dianionic adducts using ESI-MS, suggesting a reaction mechanism that begins with adduct formation, proceeds through metal atom exchange in transient dimers, and concludes with dimer dissociation into monomers. 191 Of note, intercluster reactions typically generate polydisperse product mixtures requiring subsequent separation efforts.

3.4. Synthesizing the desired ligand shell

The most straightforward way to synthesize MNCs of desired ligand shell-whether comprising single or multiple ligands—is via one-step reduction-growth of the metal-ligand complex containing the target ligand(s). 192-194 When target ligands are not commercially available, in-house synthesis provides researchers greater flexibility in ligand design, enabling the incorporation of specific functionalities or reactivities prior to complex formation with metal salts. 123,127,130

For fine-tuning the ligand shell of monodisperse MNCs, mild ligand exchange offers a versatile approach. Different from excessive ligand change, adding controlled amounts of foreign ligands can result in a multi-ligand surface with precisely regulated foreign ligand incorporation. SCXRD analysis of [Au₂₅(SR)₁₈] with two exchanged foreign ligands revealed that these foreign ligands are symmetrically bonded to the most solvent-exposed Au atoms on the surface, replacing the host thiolate ligands at the core site (Fig. 11a). 195 This observation is consistent with an associative mechanism where the ligand exchange is initiated by the association of the foreign ligand with the accessible Au atom. The site preference in ligand exchange reactions is also supported by HPLC results. 196 Adopting this approach, Liu et al. successfully introduced chirality to the ligand shell of [Au₂₃(S-c-C₆H₁₁)₁₆] via ligand exchange with chiral phosphoramidite ligands.8 To accommodate these bulky ligands, two short SR-Au(1)-SR staple motifs on the host nanocluster surface were replaced by one long SR-[Au(1)-SR]₃ motif while the Au₁₃ core structure was largely preserved despite slight twisting (Fig. 11b). In certain cases, complete ligand exchange can alter the structure of the

Review

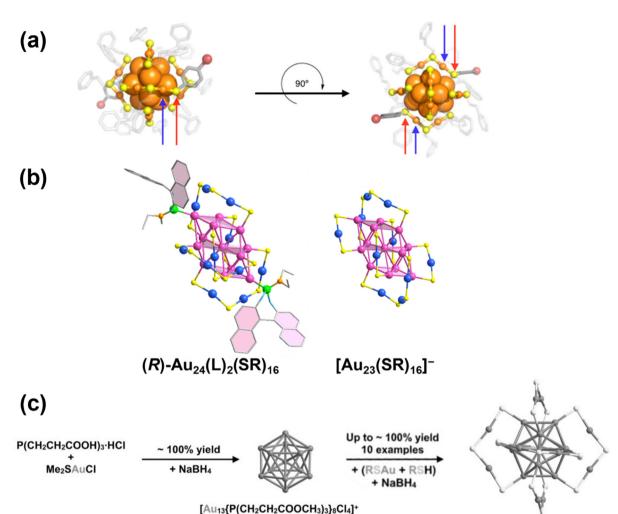


Fig. 11 (a) Structure of Au₂₅(PET)₁₈(p-BBT)₂ (p-BBT = 4-bromobenzenethiol) resolved by SCXRD. Colour label: orange, Au; yellow, S; grey, C; red, Br. Red arrows indicate the locations of exchanged p-BBT ligands and blue arrows indicate the Au atoms on the surface motif that bond to the exchanged p-BBT ligands. Reprinted with permission from ref. 195. Copyright 2014, American Chemical Society. (b) The total structures of $[Au_{23}(S-c-C_6H_{11})_{16}]^-$ after (R configuration) and before the exchange reaction with phosphoramidite. Colour label: pint, core Au; blue, motif Au; yellow, S; green, P; orange, N; light blue, O; grey, C. Reprinted with permission from ref. 8. Copyright 2023, Springer Nature Limited. (c) Schematic illustration of the reaction process of the seeded growth of [Au₂₅(SR)₁₈]⁻. Reprinted with permission from ref. 197. Copyright 2021, Wiley-VCH.

ligand shell while maintaining core integrity, as exemplified by the reversible isomerism reactions between Au₂₈(S-c-C₆H₁₁)₂₀ and Au₂₈(TBBT)₂₀.66

Seeded growth provides another effective route for ligand shell synthesis. Lei et al. developed a two-step measure for high yield synthesis of [Au₂₅(SR)₁₈] by first preparing $[Au_{13}(dppp)_4Cl_4]^+$, which possesses an identical Au_{13} core structure and can be synthesized with high yield (Fig. 11c). 197 In the second step, this Au₁₃ precursor is reacted with Au(I)-SR complexes in the presence of NaBH4 as a reducing agent to form $[Au_{25}(SR)_{18}]^-$ with yields approaching 100%. The authors observed that the ligand body can significantly affect synthesis results. When aromatic thiols are replaced with alkanethiols such as PET and S-c-C₆H₁₁, [Au₂₅(SR)₁₈] formation was prohibited, instead producing a mixture of Au₂₀(SR)₁₃ and Au₃₃(SR)₂₀. However, the detailed reaction mechanism and the role played by the ligand during the process remain to be fully elucidated.

[Au25(SR)18]-

After establishing the desired ligand shell architecture, further functionalization of ligand bodies based on their physical or chemical reactivities can be performed. As these reactions constitute crucial considerations during ligand shell design, they have been discussed extensively in Section 2.2.3 and are not repeated here.

4. Condition optimization

Optimization of reaction conditions represents the third step in synthesis planning, where parameters are systematically adjusted to achieve optimal reaction objectives. Based on the

fundamental understanding of reaction mechanisms and kinetics, scientists fine-tune influential factors that affect reaction rates and product selectivity to determine the most effective parameter combinations. Common objectives in this optimization process include maximizing reaction yield and product purity. In many cases, scientists encounter multiobjective optimization scenarios where all desired outcomes cannot be simultaneously achieved, necessitating careful tradeoff decisions to balance competing priorities while maintaining overall synthetic efficiency.

The synthesis of MNCs with well-designed core and ligand shell architectures necessitates navigation through a complex reaction landscape involving numerous intermediates and competing pathways. The selection of specific ligand and metal precursors not only determines the thermodynamically stable structures achievable but also profoundly influences the reaction kinetics through their inherent reactivity, binding affinities, and steric properties as discussed in previous sections. These precursor effects operate synergistically with other reaction condition parameters, 33,192,198 creating an optimization space of high dimensionality where strategic manipulation of conditions can control product yield and purity. This section examines the critical condition parameters that significantly impact synthesis outcomes, starting with physical parameters including reaction time, temperature and stirring condition, and thereafter focusing on chemical parameters: metal/ligand (M/L) ratio, solvent, reducing agent, pH, and use of additives (such as surfactant cations), elucidating how these factors interact with precursor chemistry to collectively determine synthesis outcomes and reviewing the optimization strategies commonly employed by researchers throughout the field.

4.1. Condition parameters at play

The synthesis of MNCs involves numerous intermediates whose evolution can be controlled through kinetic means. Condition parameters, both physical and chemical, play crucial roles during this process. In terms of physical parameters, reaction time critically influences the composition of product mixtures, thereby dictating outcomes in subsequent reaction steps. For instance, under identical conditions, a 7-day metal-ligand complex formation reaction time produced high-purity [Au₂₅(MHA)₁₈] upon subsequent NaBH₄ reduction reaction, while a 15-second reaction yielded the same product contaminated with larger nanoclusters as impurities. 199 Similarly, in the two-step synthesis of Au₂₂(SG)₁₈, a 20-minute CO incubation time was found to yield maximum purity.200

Temperature represents another fundamental physical parameter governing reaction kinetics. Through temperature variation analysis, Chen et al. revealed the endothermic nature of the reductive formation of the thermodynamically favourable [Au₂₅(MHA)₁₈] using NaBH₄, while noting that elevated temperature accelerates its decomposition. 201 The optimal condition was identified at 40 °C, which provided approximately 95% yield with favourable kinetics. In another example, Zhu et al. enhanced [Au₂₅(PET)₁₈] synthesis yield by lowering the solution temperature from room temperature to 0 °C before

thiol addition, thereby modulating the kinetics of the Au(1)-SR complex formation.²⁰² Temperature control in both complex formation and "size-focusing" thiol etching reactions enabled Zeng et al. to selectively obtain the smaller Au₄₄(TBBT)₂₈ rather than $Au_{52}(TBBT)_{32}$. 57,61

Moreover, stirring conditions, which regulate mass transfer kinetics, can significantly influence synthesis outcomes. Reducing the stirring speed during NaBH4 reduction of Au(1)-SR complexes from 400 rpm to 100 rpm altered the reaction kinetics sufficiently to shift the major product from a mixture of $10e^- \text{Au}_{39}(SR)_{29}$ and $\text{Au}_{40}(SR)_{30}$ to $4e^- \text{Au}_{24}(SR)_{20}$. Further modification combining lower stirring speed, reduced NaBH4 addition rate, and shortened reaction time yielded Au₂₀(SR)₁₆. Beyond homogenization of the reaction mixture, stirring conditions were shown to affect silver nanocluster synthesis by influencing gas-liquid mass transfer rates, particularly oxygen uptake which led to desired product formation. 204

4.1.1. Metal/ligand (M/L) ratio. With regard to chemical parameters, the metal/ligand (M/L) ratio critically determines the composition of metal-ligand complexes during MNC synthesis. ESI-MS analysis by Chen et al. revealed that high Au/SR ratios (8:1 to 3:1) produce predominantly Au chloride or hydroxide complexes such as [AuCl₂] and [H₄AuO₄] in the NaBH₄ reduction precursor solution, with thiols primarily consumed in reducing Au(III) to Au(I) and forming disulfides (Fig. 12a). 205 This insufficient protection from thiolate ligands results in the formation of larger Au nanoparticles which exhibit surface plasmon resonance. Intermediate M/L ratios (1:1 and 1:2) yield predominantly cyclic oligomeric and polymeric Au(1)-SR complexes like Au₁₀(SR)₁₀, alongside shorter chain-like complexes such as [Au₂(SR)₃], which successfully produce Au nanoclusters upon reduction. In low ratio scenarios (1:3 to 1:8), cyclic complexes are replaced by shorter species including [Au(SR)₂] and [Au₂(SR)₃], which provide enhanced protection to Au atoms and consequently exhibit decreased reactivity toward NaBH4 reduction. Despite the better protection, these short complexes arguably form short surface motifs that can only stabilize larger nanoparticles with smaller surface curvature compared to nanoclusters, ultimately leading to nanoparticle formation after sufficient reaction time. Yao et al. leveraged these smaller complexes formed at lower M/L ratios to suppress parallel aggregative growth pathways during the seeded growth of $[Au_{44}(SR)_{26}]^{2-}$, facilitating the successful synthesis of Au₃₈(SR)₂₄. 138

Additionally, ligands such as thiols function as etchants, as mentioned above, with their concentration significantly influencing etching kinetics and playing a critical role in the "sizefocusing" process. Demonstrating this principle, Liao et al. successfully synthesized the larger Au₄₉(SR)₂₇ species by introducing reduced thiol quantities to the Au_x(SR)_y etching precursor and employing shorter etching duration, in contrast to the Au₄₄(SR)₂₆ nanocluster produced when following the original protocol.206,207

4.1.2. Solvent. Solvent selection affects reaction kinetics through multiple mechanisms, including regulating reagent solubility. Lin et al. exploited solubility differences between Review

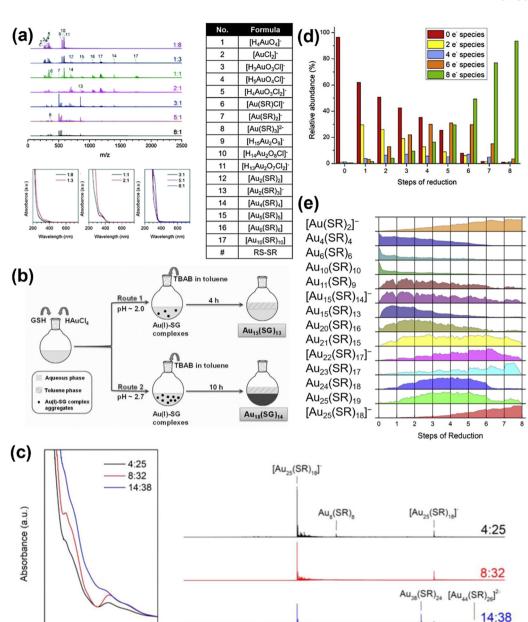


Fig. 12 (a) ESI-MS spectra of precursors formed with different SR: Au ratio and the corresponding UV-Vis absorption spectra. The species identified and labelled in ESI-MS spectra are tabulated on the right. Reproduced with permission from ref. 205. Copyright 2016, Royal Society of Chemistry. (b) Schematic illustration of the water-toluene two-phase synthesis method developed for the high-purity synthesis of Au₁₅(SG)₁₃ and Au₁₈(SG)₁₄. Reprinted with permission from ref. 209. Copyright 2013, Wiley-VCH. (c) UV-Vis absorption spectra and ESI-MS mass spectra of Au nanoclusters synthesized with different electron: Au ratios, where each mole of NaBH₄ can supply 8 electrons to each mole of Au. Reprinted with permission from ref. 142. Copyright 2018, American Chemical Society. (d) The relative abundance of species possessing different N* identified by ESI-MS after each reduction step. (e) Timedependent ESI-MS spectra showing normalized abundance of the complex and nanocluster species detected during sub-stoichiometric reduction, with the duration of each step being 30 minutes. Reprinted with permission from ref. 211. Copyright 2020, Elsevier.

2500

3000

m/z

3500

2000

MNCs by implementing a multi-solvent system comprising acetonitrile, methanol, and dichloromethane that selectively precipitated the target Cu₁₃H₁₀(SR)₃(PPh₃)₇ nanocluster to yield a highly pure product in solid form.²⁰⁸ In another approach, Yao et al. established a toluene-water two-phase system where the reducing agent borane tert-butylamine (TBAB) was

600

Wavelength (nm)

800

1500

dissolved in toluene, allowing controlled release into the reaction phase and slowing reaction kinetics to produce high-purity Au₁₅(SG)₁₃ (Fig. 12b).²⁰⁹ Notably, the authors further tuned the solubility of Au(1)-SG complex aggregates using pH to leverage the aggregation-dissociation equilibrium, demonstrating that increasing pH from 2 to 2.7 neutralized the positively charged

4000

4500

SG ligands, resulting in more aggregated Au(t)–SG complexes and reducing the TBAB/complex concentration ratio to produce $Au_{18}(SG)_{14}$.

Beyond solubility effects, the solvent environment determines the reactivity of reducing agents and the degree of thiol deprotonation. In $\mathrm{Au_{144}(PET)_{60}}$ synthesis, replacing toluene with methanol precipitates $\mathrm{Au}_x(\mathrm{SR})_y$ intermediates, preventing further growth and creating a mixture of nanoclusters with relatively narrow size distribution amenable to "size-focusing". Simultaneously, methanol promotes both NaBH₄ reduction and thiolate etching, synergistically modifying kinetics in both reduction-growth and "size-focusing" stages to yield high-purity $\mathrm{Au_{144}(PET)_{60}}$. Similar solvent-dependent outcomes were observed with $[\mathrm{Au_{37}(PPh_3)_{10}(SC_2H_4Ph)_1Cl_2]^+}$ formation in water, compared to $[\mathrm{Au_{25}(PPh_3)_{10}(SC_2H_4Ph)_5Cl_2]^{2+}}$ in ethanol. Additionally, the use of THF as solvent significantly improved the purity and yield of $\mathrm{Au_{25}(SG)_{18}}$ compared to the previously reported protocol using methanol.

4.1.3. Reducing agent. Reduction reaction kinetics can be precisely controlled through both reducing agent concentration and reducing power selection. Chen et al. employed real-time UV-Vis spectroscopy and ESI-MS characterization to identify the stoichiometric amount of NaBH4 required for synthesizing monodisperse $[Au_{25}(p-MBA)_{18}]^-$ with $[Au(SR)_2]^-$ as the only byproduct.142 Their investigation revealed that insufficient NaBH₄ resulted in under-reduction with unreacted Au₈(SR)₈, while excess NaBH4 led to over-reduction and formation of larger nanocluster impurities (Fig. 12c). Building on this stoichiometric understanding, the authors further decelerated the reaction by dividing the optimal NaBH4 amount into eight identical portions added stepwise to the reaction solution.²¹¹ According to the evolution of intermediates with distinct valence electron counts (Fig. 12d), this sub-stoichiometric approach revealed a distinct growth pathway featuring Au₂₀(SR)₁₆ as the primary 4e⁻ intermediate—differing significantly from previously reported stoichiometric NaBH4 synthesis pathways (Fig. 12e). Moreover, this method facilitated the capture of meta-stable intermediates including Au₁₁(SR)₉, $[Au_{15}(SR)_{14}]^{-}$, and $Au_{23}(SR)_{17}$, which were observable only during individual reduction steps before transforming into other stable intermediates, thereby enriching mechanistic understanding of the reduction process.

Beyond concentration manipulation, the selection of reducing agents with different reduction potentials offers another dimension for kinetic control that works synergistically with metal and ligand precursor chemistry. Wu *et al.* substituted NaBH₄ with milder TBAB, observing a gradual colour evolution from yellow to orange to black over approximately 15 minutes—a marked contrast to the rapid blackening within seconds with NaBH₄—resulting in the formation of 6e⁻ Au₁₉(PET)₁₃ rather than the thermodynamically favoured [Au₂₅(PET)₁₈]⁻.²¹² Carbon monoxide represents another mild reducing agent successfully implemented by Xie and colleagues, leading to the discovery of Au₂₂(SG)₁₈,³⁴ and high-purity synthesis of water-soluble nanoclusters of a wide size range: Au₁₅(SR)₁₃,³³ Au₁₈(SR)₁₄,³³ Au₂₀(SR)₁₆,³⁵ [Au₂₅(SR)₁₈]⁻,³³

 $[Au_{27}(SR)_{13}]^{4+},^{35}$ upon synergistic tinkering of other condition parameters such as solvent pH, solvent polarity, and use of assistant additives. Other weak reducing agents including trimethylamine borane ((CH $_3$) $_3$ N·BH $_3$) and sodium cyanoborohydride (NaBH $_3$ CN) have been effectively employed in nanocluster synthesis, each interacting uniquely with specific metal-ligand precursor combinations. 213,214

4.1.4. pH. pH adjustment through acid or base addition impacts reaction kinetics through multiple synergistic mechanisms. Beyond its influence on solubility behaviours previously discussed, 209 pH critically affects the degree of ligand deprotonation and reduction rates. Higher pH conditions promote thiol deprotonation to form thiolates that facilitate Au(III) reduction to Au(I) during complex formation, an effect supported by theoretical calculations, 198 and simultaneously accelerate thiolate-mediated etching processes. Elevated pH also decreases proton concentration which retards NaBH4 hydrolysis and consequently modifies its reduction kinetics.

Yuan et al. leveraged these mechanistic insights by adding NaOH prior to NaBH₄ introduction, achieving high-purity synthesis of mono- and multi-thiolate protected [Au₂₅(SR)₁₈] through precise balancing of reduction and etching kinetics. 192 Conversely, Wu and colleagues employed acidic conditions, via nitric acid addition, to promote reduction while retarding thiolate etching, leading to successful synthesis of novel Au₅₂(PET)₃₂ nanoclusters that are isomeric to previously discovered Au₅₂(TBBT)₃₂.²¹⁵ Notably, as acid introduction preceded thiol addition in their protocol, complex formation kinetics were altered as well, contributing to the formation of Au₅₂(PET)₃₂. Acid can be strategically introduced at different stages of the synthesis process to modulate specific reaction steps. Wu's group demonstrated this versatility by adding acid before NaBH₄ during Au₄₂(TBBT)₂₆ synthesis, ²¹⁶ while Shichibu et al. incorporated acid at the beginning of the "size-focusing" step when working with Au_N(dppe)_xCl_y precursors to produce phosphine-protected [Au₁₃(dppe)₅Cl₂]³⁺.82 When considering CO as the reducing agent, reaction kinetics are promoted by hydroxide ions, resulting in the formation of larger nanoclusters at higher pH values (Fig. 13a). 33 These observations collectively demonstrate that pH serves as a powerful parameter for tuning nanocluster synthesis through its multifaceted effects on ligand behaviour and reducing agent performance, allowing researchers to navigate complex reaction landscapes with greater precision.

4.1.5. Use of additives. Additives that interact with ligands via intermolecular interactions provide another powerful approach for tuning reaction kinetics. Surfactant cations with long hydrocarbon tails represent the most commonly employed additives, primarily serving to transfer aqueous metal ion precursors to organic phases where subsequent reactions occur. However, Jin and colleagues discovered that TOA^+ were indispensable for successful one-phase synthesis of $Au_{144}(SR)_{60}$, even when phase transfer was not required. This likely stems from specific interactions between TOA^+ and 1-hexanethiolate ligands that influence the structures and compositions of initial Au(I)–SR polymers and mixed-size $Au_x(SR)_y$

Review

pH7 Au₁₀₋₁₂SR₁₀₋₁₂ (a) glutathione pH9 Au₁₅SR₁₃ pH adjusting + CO-reduction pH10 Au(I)-SR complexes Au₁₈SR₁₄ Au₂₅SR₁₈ (b) Short Au(I)-SR Complexes 1-Octyl-3-methylimidazolium [Au25(SR)18] (OMIm+) CO **lonic Liquid** (IL) Assisted **Size Control** [Au₂₀(SR)₁₆]⁰

Fig. 13 (a) Schematic illustration of the facile synthesis of GSH-protected Au nanoclusters of various sizes by adjusting the pH value through the COreduction approach. Reprinted with permission from ref. 33. Copyright 2013, American Chemical Society. (b) Schematic illustration of the tailoring of Au nanocluster sizes via the introduction of duel electrostatic π - π stacking interactions between ionic liquid cations such as OMIm⁺ and deprotonated p-MBA. Reprinted with permission from ref. 35. Copyright 2021, Wiley-VCH.

π-π Enhanced

Ion-Pairing

Long Au(I)-SR

Complexes With OMIm *

intermediates, thereby facilitating the formation of Au₁₄₄(SR)₆₀ after subsequent treatments-though detailed mechanistic studies are still underway.

Extending this approach, Zhu et al. introduced ionic liquid cations such as 1-octyl-3-methylimidazolium (OMIm+) to interact with the deprotonated p-MBA ligands through duel electrostatic π - π stacking interactions (Fig. 13b).³⁵ As a result, Au₂₀(SR)₁₆ is stabilized by the altered reaction kinetics. By systematically varying the side chain length of the cation while simultaneously adjusting other parameters (cation concentration, pH, and solvent relative permittivity), the researchers demonstrated remarkable control over product selectivity: C12 chain length cation produced the smaller Au₁₅(SR)₁₃ nanocluster, C6 cation yielded [Au₂₅(SR)₁₈]-, and C4 cation generated

the larger [Au₂₇(SR)₁₃]⁴⁺. A similar reaction-directing effect can be induced by metal ions, as demonstrated by the essential role of Cd2+ in synthesizing the non-fcc-structured isomer of Au₄₂(TBBT)₂₆. 217

Small sodium metal salts have been found to accelerate isoelectronic conversion between [Au₂₃(p-MBA)₁₆] and [Au₂₅(p-MBA)₁₆] MBA)₁₈]^{-.166} The degree of kinetic enhancement correlates positively with ionic strength (determined by concentration or anion valency) but remains independent of anion identity. This phenomenon is attributed to the weakening of electrostatic repulsion between negatively charged Au(1)-SR complexes and $[Au_{23}(p-MBA)_{16}]^-$ resulting from compression of the electric double layer. This finding illustrates how even simple ionic additives can significantly modulate reaction kinetics through

electrostatic effects, providing researchers with additional tools for navigating complex nanocluster synthesis landscapes.

4.2. Condition optimization strategies

The one-factor-at-a-time (OFAT) approach represents the most commonly adopted optimization strategy in metal nanocluster synthesis. This conventional method involves identifying all relevant parameters and optimizing them sequentially, beginning with a parameter selected by researcher's decision.²¹⁸ In MNC synthesis, OFAT is the strategy of choice in the optimization of various parameters discussed above, including ligand choice, 102 M/L ratio, 32,205 solvent, 31 reducing agent, 142 pH, 33 and use of additives. 166 However, as these parameters exhibit significant synergistic effects, achieving optimal synthesis protocols in terms of yield or purity typically requires fine-tuning of multiple parameters simultaneously. Given OFAT's inherent limitations in efficiency and accuracy when navigating multidimensional design spaces, researchers are turning to global optimization strategies for MNC synthesis, which will be discussed in detail in Section 5.2.

5. Opportunities offered by datadriven approaches

As discussed above, the synthesis planning of MNCs inherently relies on complex structure-property (target design), precursorproduct (route development) and condition-objective/product relationships (condition optimization). However, the construction of these crucial relationships is challenging. First, obtaining precise structural information can be experimentally difficult, particularly when the surface ligand is highly flexible (like water-soluble ligand-capped MNCs), ideal product purity for crystallization is difficult to achieve, and solvent-dissolved MNCs exhibit unique surface dynamics. While computational simulations provide theoretical evidence in these cases, the structural complexity of MNCs demands substantial computational resources. Second, the multidimensional parameter space created by interconnected reaction precursors and various condition parameters complicates the study of synthesis-product relationships using conventional methods. Third, most established relationships—such as core growth templates and condition parameter effects-remain largely qualitative, derived from limited datasets and semi-analytical theories with restricted applicability across the broader MNC design space.

Data-driven methodologies offer promising solutions to these challenges by providing a quantitative framework where statistical models are constructed using carefully selected input and output descriptors that capture essential information about the chemical structure, properties, and reaction conditions. In this approach, complex relationships are abstracted as interactions among descriptor combinations. While property characterization data and reaction condition parameters typically serve as quantitative descriptors, chemical species and structural representations can be characterized through compositional, topological, structural, quantum-chemical, or mathematical descriptors. By integrating advanced algorithms with high-throughput data generation, this approach enables robust predictions based on high-dimensional relationship modelling with reliable accuracy, while facilitating straightforward model updates and adaptations.219,220

This section explores how data-driven methodologies can advance MNC synthesis planning through structure prediction

Table 1 Summary of works adopting data-driven approaches for MNC rational design and synthesis

Objectives	Methods adopted	Examples
Structure prediction acceleration	Random forest	Prediction of CO adsorption on Ag-doped Au nanoclusters ²²¹
	Convolutional neural networks	Hydride location prediction ^{222,223}
	Feedforward neural networks with simulated annealing	Prediction of ${\rm Au}_{25}$ -protein interactions ²²⁴
	Distance-based ML	${ m Au_{38}(SR)_{24}}$ structure prediction at varied temperatures ^{22.5}
	Local search algorithm	Structure prediction of metal-ligand interfaces of Au/Ag based MNCs ²²⁶
	Recurrent neural networks	PL property prediction of hairpin-DNA templated Ag nanoclusters ²²⁷
Global condition optimization	Siamese neural networks with graph convolutional neural networks	Au ₂₅ synthesis outcome prediction ⁴⁶
	Random forest with HTE	Prediction of synthesis yield for Au–Cu bimetallic nanoclusters ²²⁸
	Convolutional neural networks	Composition prediction from UV-Vis absorption data ²²⁹
Synthesis-property relationship modelling	Support vector machine with HTE	Precursor design for desired DNA-Ag nanocluster PL emission colour ^{230,231}
	Variational autoencoder with HTE	Precursor design for desired DNA–Ag nanocluster PL colour and brightness ⁴⁷
	Extreme gradient boosting regressor	Synthesis of highly luminescent glutathione- protected Au nanoclusters ²³²

acceleration, global condition optimization modelling, direct synthesis-property relationship modelling, and systematic model generalization and adaptation-facilitating more efficient and precise design and synthesis strategies (Table 1).

5.1. Structure prediction acceleration

When experimental determination of precise structures proves challenging, computational structure prediction provides theoretical insights that rationalize macroscopic properties of MNCs and facilitate structure-property relationship development. For example, combined molecular dynamics (MD) simulations and density functional theory (DFT) calculations have revealed dynamic interactions at the ligand-metal interface of water-dissolved m-MBA-protected Au₆₈, explaining its simultaneous high stability and reactivity in protein conjugation reactions.²³³ Similarly, MD simulations identifying specific binding sites mediated by electrostatic interactions and hydrogen bonds have provided microscopic elucidation of stoichiometric binding between bovine serum albumin (BSA) and [Au₂₅(SR)₁₈]⁻, advancing the understanding of MNC-protein interactions for bio-related applications. 124 However, these simulation processes typically require intensive computational resources due to the structural complexity of MNCs.

ML approaches have demonstrated remarkable efficacy in accelerating structure prediction processes. For example, a ML model trained on descriptors derived from adsorbate-free and

nonrelaxed structures has enabled rapid filtering of potential candidates based on predicted CO adsorption energy on Agdoped Au nanoclusters, significantly accelerating the research process compared to exclusive reliance on time-consuming DFT calculations. 221 Notably, this model performed with good accuracy for larger Au nanoclusters absent from the training data, demonstrating its generalizability. In another example, Wang et al. employed CNNs trained on existing structure libraries of hydride-doped Cu nanoclusters to model relationships between local chemical environments and hydride occupancy probability. 222 Combining with symmetry constraint considerations, the authors significantly reduced the number of potential structural candidates requiring DFT optimization. However, these prediction results still require experimental verification to confirm the accuracy and validity of these ML models.

ML approaches have also advanced simulations of MNCprotein interactions. Addressing challenges including the lack of suitable force fields and the wide range of simulation timescales required for conventional modelling of dynamic MNCprotein interactions, Pihlajamäki et al. developed a ML methodology comprising a feedforward neural network (FNN) that predicts Coulomb and van der Waals contributions to interaction energies in Au nanocluster-protein complexes at a coarse-grained level. 224 This prediction is followed by optimization via Monte Carlo-based structure search and refinement to atomic-scale structures. The researchers developed graph

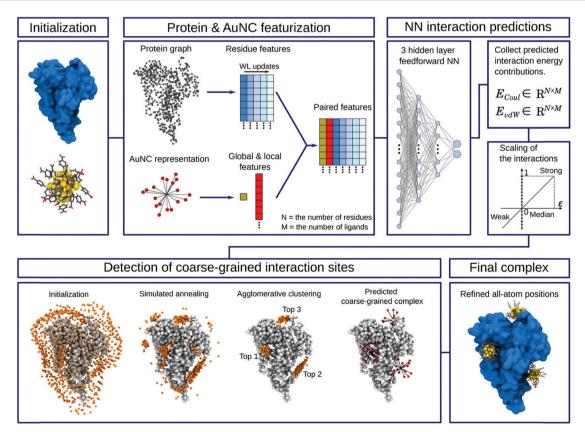


Fig. 14 The schematic illustration of the ML integrated framework for nanocluster-protein interaction simulations. Reprinted with permission from ref. 224. Copyright 2024, Wiley-VCH

theory-based representations of Au nanoclusters and protein structures as descriptors, effectively simplifying structural complexity while preserving essential information (Fig. 14). The method was validated through subsequent MD simulations, demonstrating robust and accurate predictions of preferred binding sites between proteins and MNCs-even for larger proteins and MNCs absent from the training data. Analysis of predicted binding sites revealed the significance of electrostatic interactions between positively charged protein residues, particularly lysine (LYS) and ARG, and negatively charged MNC ligands (such as p-MBA). These results establish this method as a facile and reliable tool for studying MNC-protein interactions and dynamic properties of nano-bio interfaces at atomic scale, overcoming computational limitations of traditional approaches. However, the model could be potentially improved by incorporating hydrogen bonding when predicting interaction energies between nanoclusters and proteins, as hydrogen bonds play crucial roles in the binding of the Au₂₅(p-MBSA)₁₈-BSA (p-MBSA = para-mercaptobenzenesulfonic acid) complex alongside electrostatic interactions. 124 This may contribute to the model's limited performance in predicting the binding behaviour of the $(Au_{25}(p-MBSA)_{18})-HSA$ (HSA = human serum albumin) complex.

When high-quality datasets are already available, a statistical modelling-based approach enables rapid structure predictions with accuracy close to the quantum mechanical level. For instance, a model trained to correlate the local atomic environment of hydrogen in Cu- and Pd-doped [Au₂₅(SR)₁₈] with the hydrogen-nanocluster interaction energy for catalyst design in electrocatalytic hydrogen evolution reactions achieved predicted energies within 0.1 eV difference compared to DFT calculation results.234

5.2. Modelling for global condition optimization

The multidimensional nature of synthesis parameters poses significant challenges for conventional optimization approaches in MNC synthesis. Traditional OFAT methodologies, which optimize condition parameters sequentially (Fig. 15a), demonstrate inherent limitations in identifying global optima and elucidating synergistic relationships among interdependent parameters.²¹⁸ Furthermore, the expansive design space renders comprehensive experimentation across all possible parameter combinations prohibitively resource-intensive. Statistical modelling approaches, therefore, offer a methodologically robust alternative that effectively balances the critical trade-off between experimental efficiency and optimization accuracy.

Li et al. pioneered the application of statistical modelling to predict MNC synthesis outcomes (success or failure) based on specific parameter sets. 46 Their innovative approach employed a Siamese neural network (SNN) stacked with a graph convolutional neural network (GCNN) classification model, trained to predict whether particular parameter combinations would yield monodisperse [Au₂₅(SR)₁₈] protected by thiols with diverse ligand bodies. After training, the model was deployed to predict synthesis outcomes for all parameter combinations within the design space, enabling the construction of a decision tree to provide simple guidance on subsequent nanocluster synthesis, albeit with acknowledged precision limitations (Fig. 15b). Extending this data-driven methodology to bimetallic systems, Tang et al. developed a condition-yield relationship model utilizing the random forest algorithm. 228 The team constructed a HTE platform capable of conducting up to 264 simultaneous reactions to facilitate data collection. The model exhibits good prediction accuracy for samples with <50% yield but low accuracy in the >50% region. To further improve on such models, additional data collection in less represented regions is required through an iterative process of data collection, model updating, and experiment suggestion—a methodology known as active learning.

Active learning methodologies have demonstrated remarkable success in organic synthesis optimization, notably through Bayesian optimization algorithms. 40 In a typical optimization

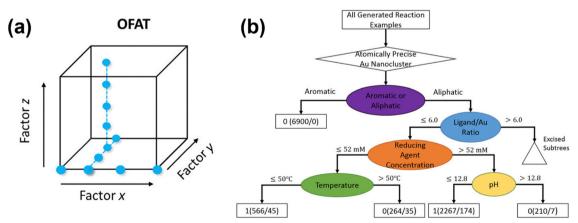


Fig. 15 (a) Schematic illustration of the OFAT optimization strategy for parameter space exploration, where each dot represents an experiment. Reprinted with permission from ref. 218. Copyright 2023, American Chemical Society. (b) Decision tree generated by statistical modelling for the prediction of $[Au_{25}(SR)_{18}]^-$ synthesis outcome in the aqueous phase, where ovals represent decision nodes and rectangles represent reaction-outcome bins (where the reaction outcome is represented as 1 for success and 0 for failure and number of reaction examples correctly and incorrectly classified are denoted in parentheses before and after the slash respectively). Triangles depict excised subtrees due to both extra small examples in that branch and chemical intuition. Reprinted with permission from ref. 46. Copyright 2019, Wiley-VCH.

process, a statistical surrogate model (e.g., Gaussian process) is initially constructed using a set of preliminary experimental data, mapping the design space with expected mean values and variances. An acquisition function (e.g., expected improvement, which typically prioritizes conditions with the highest expected improvement compared to the current best) then guides the selection of subsequent experiments for model refinement. To avoid convergence on local optima, the algorithm strategically balances between exploiting areas of high predicted performance and exploring regions with high uncertainty. This iterative process continues until identifying the globally optimal parameter set. By modifying either the model output or the acquisition function, optimization can address multiple objectives simultaneously. 219,235 HTE platforms can significantly accelerate this data collection process, enhancing efficiency. Upon successful optimization, the resulting surrogate model trained with comprehensive experimental data-provides an accurate representation of condition-objective relationships, enabling clear visualization of synergistic effects among reaction parameters through response surface modelling and reaction kinetics and mechanistic analysis facilitating (Fig. 16a).

Exploration through the design space beyond single-product regions enables the construction of comprehensive conditionproduct relationships, providing deeper insights into the roles of various synthesis parameters. Du et al. demonstrated this approach in polyoxometalate synthesis, where phase diagrams derived from design space exploration and condition optimization revealed the complex interplay among reaction parameters in crystal formation and intermediate linker length determination, ultimately dictating product identity. 236 This methodology is particularly relevant for MNC synthesis, where many nanoclusters are produced through kinetically controlled processes in which minor parameter adjustments can significantly alter product outcomes. Well-constructed phase diagrams reveal valuable kinetic and mechanistic information while providing clear guidance for protocol optimization and rational design strategies.

The overlapping regions in the phase diagram, where multiple kinetically controlled species coexist, present opportunities for advanced kinetic modelling approaches. Utilizing ML algorithms, Li et al. demonstrated the possibility of predicting the composition of a MNC mixture solution and the relative abundance of the respective compositional species in the ESI-MS spectrum from UV-Vis absorption results (Fig. 16b).²²⁹ As such, once a model is well-trained with in situ UV-Vis absorption and real-time ESI-MS measurements, it is technically possible to reduce the reliance on the labour-intensive time-course ESI-MS measurements while still enabling compositional evolution monitoring for kinetic model development. Such models would allow prediction the relative abundance of byproducts under similar reaction conditions, enabling synthetic route ranking based on byproduct profiles in optimized synthesis products. This information becomes invaluable during the route development stage of the synthesis planning process, facilitating more informed decision-making based on quantitative predictions rather than qualitative assessments.

5.3. Synthesis-property relationship modelling

When the product structural information is difficult to determine due to either polydispersity or difficulty in obtaining single crystals, rational MNC target design can instead be anchored to reaction precursors or synthetic condition parameters to construct direct synthesis-property (in terms of precursor- or condition-property) relationship models. The construction of these high-dimensional relationships has become increasingly feasible through statistical modelling approaches.

The precursor-property relationship modelling approach has demonstrated notable success in designing DNAstabilized Ag nanoclusters with targeted fluorescence properties, even without precise structural characterization. Copp et al. demonstrated this approach by training a classification model using Ag nanocluster products synthesized with 1432 distinct DNA oligomers to predict product colour, categorized by fluorescence spectra peak wavelength (Fig. 17a). 230 In this system, the DNA nucleobase sequence determines the cluster size and, consequently, its photoluminescence characteristics. The researchers parameterized DNA sequences as arrays of approximately 120 binary descriptors, each representing the presence or absence of specific base patterns in the DNA oligomer. This model successfully generated DNA sequences for synthesizing products in the desired "green" and "very red" colour classes with selectivity enhancements of 81% and 330%, respectively, compared to the training data. The authors noted that many products targeting green emission exhibited brightness below the detection threshold, contributing to relatively low selectivity in this category.

To enhance the precision and accuracy of this precursorproperty relationship model, the same research group incorporated chemical insights obtained from the recently resolved crystal structure of DNA-stabilized Ag₁₆. ¹⁴⁷ This advancement enabled refinement of the feature engineering protocol to include both adjacent and nonadjacent base patterns, accounting for the three-dimensional relationship between the Ag core and the DNA ligand. 231 Based on this improved methodology, the authors increased the selectivity by 12.3 times targeting bright near-infrared nanoclusters with promising deep tissue bioimaging capabilities. Furthermore, to address the challenge of insufficient brightness in some products, the team upgraded their model to enable simultaneous colour and brightness selection, which successfully reduced the occurrence of products with inadequate brightness while increasing the selectivity of those with both desired colour and brightness characteristics.47 Through this stepwise approach, the precursor-property relationship model was progressively refined to guide target design with increasingly specific performance requirements. While the current model successfully generates DNAs for nanoclusters above a brightness threshold, future enhancements could focus on ranking DNA candidates based on predicted brightness values. This advancement would not

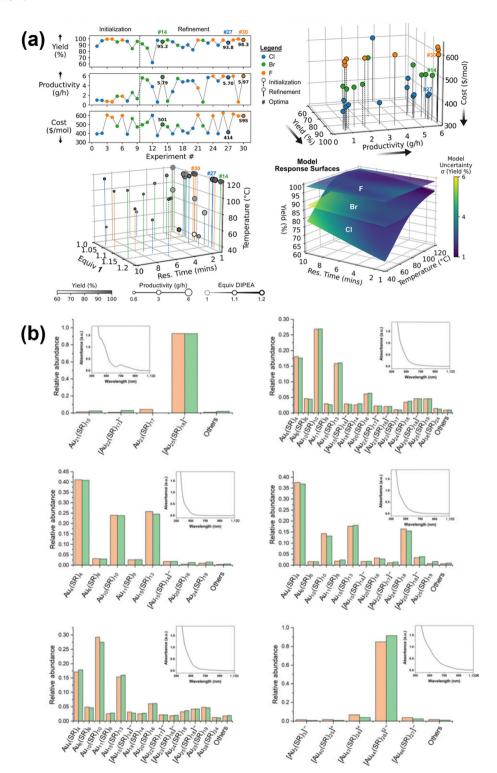


Fig. 16 (a) Graphs depicting the objective values versus experiment number (top left), three-dimensional plots of objective values (top right) and condition parameter values experimented (bottom left), and the response surface modelling the relationship between yield and condition parameters generated by the Gaussian process models trained with experimental data (bottom right) after a multi-objective organic synthesis optimization campaign. Reprinted with permission from ref. 219. Copyright 2022, American Chemical Society. (b) Predicted relative abundance (green bars) based on UV-Vis absorption spectra (shown in insets) and actual relative abundance (orange bars) of the compositions in polydisperse Au nanocluster samples. Reprinted with permission from ref. 229. Copyright 2023, Springer Nature Limited.

only facilitate the identification of nanoclusters with the highest emission intensity-desirable for bioimaging and biosensing applications—but also reduce the candidate pool size, thereby enhancing the efficiency of identifying suitable

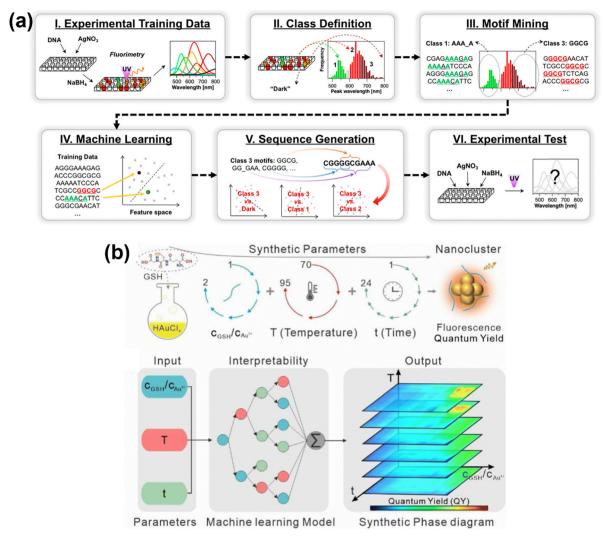


Fig. 17 (a) Schematic illustration of the methodology for designing DNA-stabilized Ag nanoclusters with specific colour classes via DNA sequence design using ML algorithms. Reprinted with permission from ref. 230. Copyright 2018, American Chemical Society. (b) Schematic illustration of the synthesis of GSH-protected photoluminescent Au nanoclusters (top) and the workflow for the construction of the synthetic phase diagram by ML algorithms. Reproduced with permission from ref. 232. Copyright 2023, Royal Society of Chemistry.

DNA sequences for potential highly performing metal nanocluster candidates.

Similarly, condition-property relationship modelling has proven valuable for designing photoluminescent glutathionestabilized Au nanoclusters. Due to the significant roles played by the various condition parameters in the synthesis of photoluminescent glutathione-stabilized Au nanoclusters and the unavailability of precise structural information, Jin et al. developed a condition-property relationship model for target design to optimize QY (Fig. 17b).232 Their investigation focused on a three-dimensional design space comprising M/L ratio, reaction temperature and synthesis duration. The model's accuracy was validated by low error between the experimental and predicted QY values at 10 randomly selected points in the design space. Analysis of the trained model revealed complex parameter interactions, with high QY products obtained under specific conditions: low M/L ratio (0.5-0.63), short reaction time (3.3–10.2 h) and high temperatures (85–95 °C). These insights enabled the researchers to successfully reverse-engineer highly photoluminescent glutathione-stabilized Au nanoclusters with optimized properties. However, the model can be further enhanced by incorporating additional condition parameters such as the solvent used and the pH of the solution to cover a higher proportion of the entire synthesis design space.

These direct modelling approaches significantly facilitate the design of target MNCs whose structural information is difficult to obtain or whose monodispersity is challenging to achieve, thereby circumventing the labour-intensive process of structural determination for numerous less qualified MNC candidates. By establishing quantitative relationships between readily controllable synthesis parameters and desired functional properties, these methodologies provide an efficient pathway for rational MNC design in cases where traditional structure-guided approaches are not readily available.

5.4. Model generalization and adaptation

Current principles and models in metal nanocluster (MNC) synthesis planning are typically restricted to specific MNC subgroups due to limited dataset sizes, constraining predictability for new systems. 57,232 This reflects a fundamental tradeoff between model complexity and generalizability: while complex models can capture intricate synthesis relationships essential for accurate MNC prediction, they require extensive data to generalize effectively and risk overfitting in new chemical spaces with limited training examples. Conversely, simpler models may generalize more readily but may miss critical nonlinear relationships governing synthesis outcomes.

Data-driven approaches strategically navigate this trade-off through generalization techniques that enable the deployment of appropriate complex models while maintaining broad applicability. 46,225,226,231 The effectiveness of model adaptation depends primarily on the chemical similarity between source and target domains. When similarity is high, transfer learning provides a powerful framework for leveraging knowledge from data-rich domains to guide synthesis in related but data-limited domains.^{237,238} This transfer can be implemented through several complementary approaches: instance-based methods reweight samples (such as species or ligand molecules) to account for domain differences; feature-based approaches utilize descriptors exhibiting minimal numeric variations between domains; parameter-based approaches directly transfer model parameters (such as regression model coefficients) or chemical insights; and relational-based methods map connections between different chemical spaces to apply prior knowledge.

For example, models initially developed for condition-purity relationships of a specific MNC species can be systematically expanded along precursor dimensions through feature-based transfer learning. This expansion enables researchers to probe the complex roles of different precursor types (such as ligand) in synergism with the condition parameters during synthesis while preserving established relationships. Furthermore, by establishing correlations between synthesis condition parameters, reaction precursors, and product identity and purity, researchers can elucidate the complex synthesis design space of MNCs (e.g., MNC size and structural dependencies on ligand and metal precursors) and develop comprehensive synthesis protocols with systematic understanding of parameter interactions.

For systems characterized by low domain similarity, combining transfer learning with active learning offers iterative performance refinement through strategic data incorporation. 220,239 These updates can integrate both literature-derived information and in-house experimental results, though data quality remains critical-requiring representative sampling and consistent descriptor usage across datasets. While literature data integration faces challenges including publication bias toward successful reactions and inconsistent descriptors, HTE platforms provide an effective solution by efficiently generating systematic, unbiased datasets in-house, specifically designed to improve model adaptability across diverse MNC systems. These datasets can be deliberately structured to address the gaps in existing knowledge

bases, enhancing the robustness of predictive models across chemical space.

The continued development of these modelling efforts, coupled with coordinated data collection strategies, will progressively build a comprehensive MNC data library shared throughout the research community. This collaborative approach facilitates the development of increasingly general and reliable predictive models for precise MNC design, ultimately bridging the gaps between targeted properties, MNC structures and optimal synthesis conditions. As these models evolve to incorporate a broader range of MNC systems, they will enable more rapid adaptation to emerging research directions and accelerate the discovery of novel MNCs with tailored functionalities.

Conclusion and outlook

In summary, this perspective has examined the fundamental considerations in the rational design and synthesis of MNCs within a comprehensive synthesis planning framework. We have systematically explored the design principles governing target core and ligand shell composition and architectures, critical factors in synthetic route selection, and the profound effects of various condition parameters on successful MNC synthesis outcomes. Data-driven approaches offer transformative opportunities to enhance both efficiency and precision throughout the synthesis planning process and have demonstrated capabilities in accelerating structure prediction to deepen our understanding of complex MNC structures and establishing quantitative relationship models that provide predictive capabilities beyond traditional qualitative approaches. Moreover, model generalization and adaptation through transfer learning and active learning methodologies enables knowledge transfer from priorly studied domains to new target problems. However, several challenges must be addressed for effective data-driven approach adoption in MNC synthesis planning.

Firstly, the existence of uncertainties arising from both noisy training data and model ignorance can affect model performance. Noisy data pose difficulties in model training that result in low accuracy and unreliable predictions. Effective data preprocessing—including smoothing for spectral data,240 and feature engineering techniques such as normalization and dimensionality reduction—can mitigate aleatoric uncertainty effects. Algorithm selection critically impacts uncertainty handling. While Gaussian-based models intrinsically account for uncertainties, approaches like autoencoders provide dimensionality reduction benefits. Besides, uncertainty quantification techniques such as ensemble-based methods which involve training multiple model replicates and calculating predictions as arithmetic means, 241 and mean-variance estimation (MVE) incorporating additional variance prediction neurons can be incorporated for more robust and reliable predictions.²⁴² However, further research is needed to identify optimal approaches for specific problems. These techniques

enhance prediction confidence and enable informed decisionmaking while providing foundations for active learning-based model refinement.

Secondly, beyond relationship modelling and prediction, model explainability should be addressed to reveal underlying prediction mechanisms. Post-hoc explanation methods such as local interpretable model-agnostic explanations (LIME) and shapley additive explanations (SHAP) provide model-agnostic tools. 243,244 For example, SHAP analysis identified gas hourly space velocity and temperature as critical features for higher alcohol synthesis catalyst development.245 However, SHAP treats features independently, ignoring causal relationships. Partial dependence plots can probe synergistic feature interactions when informative descriptors are employed.²⁴⁶ For deep learning models, attention mechanisms highlight crucial input data regions affecting predictions.²⁴⁷ The incorporation of these techniques can not only enhance prediction reliability, but also reveal synthetic and mechanistic insights that guide future research directions.

The foundation of successful modelling approaches relies on high-quality data collection. It is crucial to develop unbiased datasets with descriptors carefully designed not only for the specific relationships under investigation but also selected with foresight toward their utility in future model generalization and cross-domain adaptability. HTE significantly enhances efficiency by enabling systematic exploration of parameter spaces that would be prohibitively time-consuming through traditional methods—a capability particularly valuable when confronting the vast synthetic design space inherent in MNC systems.²⁴⁸ Standardized reporting protocols for experimental design, characterization, data collection, featurization, and model training should be promoted to facilitate data sharing, enhance reproducibility, and establish collective databases. 249-251 Such practices alleviate data preprocessing workloads, enhance efficiency through facilitating automated literature data mining using ML tools including large language models (LLM), 252-254 and provide high-quality data for robust model training.

Looking forward, the continued expansion of the MNC library, accompanied by robust property characterization data, will progressively enrich comprehensive MNC databases, enabling adoption of powerful yet data-demanding tools such as generative models for inverse design of novel materials with desired properties. 255,256 These approaches have demonstrated promising capabilities across diverse fields, from crystalline inorganic materials generation under chemistry or property constraints,²⁵⁷ to applications in organic molecules,²⁵⁸ drug discovery, ²⁵⁹ metal-organic frameworks (MOFs), ³⁶ and plasmonic nanoparticles.260 However, ensuring synthesizability of computationally predicted materials remains a critical challenge requiring substantial research effort. This growing knowledge repository allows for increasingly sophisticated structureproperty relationship modelling, advancing the field toward the ambitious goal of "dial-a-MNC" capabilities where on-demand MNCs can be synthesized with precisely tailored properties. Through strategic integration of computational approaches

with experimental advances, the rational design and predictive synthesis of metal nanoclusters with tailored properties for specific applications becomes increasingly achievable, opening new frontiers in nanomaterial science and technology.

Author contributions

J. Xie supervised this work. J. Xie and J. Lyu conceived the idea and wrote the manuscript. J. Lyu conducted the literature survey. J. Qian and J. Lyu participated in visualization. J. Qian and Z. Yang commented on the manuscript.

Conflicts of interest

There are no conflicts to declare.

Data availability

No primary research results, software or code have been included, and no new data were generated or analysed as part of this review.

Acknowledgements

We acknowledge the Ministry of Education, Singapore, for their financial support, through the Academic Research Fund (AcRF) Grant No. A-8002406-00-00. We would also like to acknowledge the support from the National Natural Science Foundation of China (22071174).

Notes and references

- 1 T. J. Struble, J. C. Alvarez, S. P. Brown, M. Chytil, J. Cisar, R. L. DesJarlais, O. Engkvist, S. A. Frank, D. R. Greve, D. J. Griffin, X. Hou, J. W. Johannes, C. Kreatsoulas, B. Lahue, M. Mathea, G. Mogk, C. A. Nicolaou, A. D. Palmer, D. J. Price, R. I. Robinson, S. Salentin, L. Xing, T. Jaakkola, W. H. Green, R. Barzilay, C. W. Coley and K. F. Jensen, J. Med. Chem., 2020, 63, 8667-8682.
- 2 M. Zhu, C. M. Aikens, F. J. Hollander, G. C. Schatz and R. Jin, J. Am. Chem. Soc., 2008, 130, 5883-5885.
- 3 H. Lin, X. Song, O. J. H. Chai, Q. Yao, H. Yang and J. Xie, Adv. Mater., 2024, 36, 2401002.
- 4 Z. Luo, X. Yuan, Y. Yu, Q. Zhang, D. T. Leong, J. Y. Lee and J. Xie, J. Am. Chem. Soc., 2012, 134, 16662-16670.
- 5 N. Goswami, Q. Yao, Z. Luo, J. Li, T. Chen and J. Xie, J. Phys. Chem. Lett., 2016, 7, 962-975.
- 6 Z. Wu, Q. Yao, O. J. H. Chai, N. Ding, W. Xu, S. Zang and J. Xie, Angew. Chem., Int. Ed., 2020, 59, 9934-9939.
- 7 Z. Liu, L. Luo and R. Jin, Adv. Mater., 2024, 36, 2309073.
- 8 C. Liu, Y. Zhao, T.-S. Zhang, C.-B. Tao, W. Fei, S. Zhang and M.-B. Li, Nat. Commun., 2023, 14, 3730.
- 9 I. Dolamic, S. Knoppe, A. Dass and T. Bürgi, Nat. Commun., 2012, 3, 798.

- 10 Y. Li, H. K. Kim, R. D. McGillicuddy, S.-L. Zheng, K. J. Anderton, G. J. Stec, J. Lee, D. Cui and J. A. Mason, J. Am. Chem. Soc., 2023, 145, 9304-9312.
- 11 C. Zhu, Z.-L. Chen, H. Li, L. Lu, X. Kang, J. Xuan and M. Zhu, J. Am. Chem. Soc., 2024, 146, 23212-23220.
- 12 S. Masuda, K. Sakamoto and T. Tsukuda, Nanoscale, 2024, 16, 4514-4528.
- 13 F. Xiao, Y. Chen, J. Qi, Q. Yao, J. Xie and X. Jiang, Adv. Mater., 2023, 35, 2210412.
- 14 K. Zheng and J. Xie, Acc. Mater. Res., 2021, 2, 1104-1116.
- 15 G. Yang, Z. Wang, F. Du, F. Jiang, X. Yuan and J. Y. Ying, J. Am. Chem. Soc., 2023, 145, 11879-11898.
- 16 Y.-S. Chen and P. V. Kamat, J. Am. Chem. Soc., 2014, 136, 6075-6082.
- 17 Y. Li, M. Zhou and R. Jin, Adv. Mater., 2021, 33, 2006591.
- 18 Y. Niihori, S. Miyajima, A. Ikeda, T. Kosaka and Y. Negishi, Small Sci., 2023, 3, 2300024.
- 19 Q. Yao, M. Zhu, Z. Yang, X. Song, X. Yuan, Z. Zhang, W. Hu and J. Xie, Nat. Rev. Mater., 2025, 10, 89-108.
- 20 B. Zhang, J. Chen, Y. Cao, O. J. H. Chai and J. Xie, Small, 2021, 17, 2004381.
- 21 F. Bertorelle, K. D. Wegner, M. Perić Bakulić, H. Fakhouri, C. Comby-Zerbino, A. Sagar, P. Bernadó, U. Resch-Genger, V. Bonačić-Koutecký, X. Le Guével and R. Antoine, Chem. -Eur. J., 2022, 28, e202200570.
- 22 J.-H. Huang, H. Zhang, Z.-Y. Wang, J.-H. Hu, J. Li, J. Cai and S.-Q. Zang, J. Am. Chem. Soc., 2025, 147, 16593-16601.
- 23 R. Jin, C. Liu, S. Zhao, A. Das, H. Xing, C. Gayathri, Y. Xing, N. L. Rosi, R. R. Gil and R. Jin, ACS Nano, 2015, 9, 8530-8536.
- 24 H. Shan, J. Shi, T. Chen, Y. Cao, Q. Yao, H. An, Z. Yang, Z. Wu, Z. Jiang and J. Xie, ACS Nano, 2023, 17, 2368-2377.
- 25 R. Jin, C. Zeng, M. Zhou and Y. Chen, Chem. Rev., 2016, **116**, 10346–10413.
- 26 T. Chen, H. Lin, Y. Cao, Q. Yao and J. Xie, Adv. Mater., 2021, 34, 2103918.
- 27 J. Qian, Z. Yang, J. Lyu, Q. Yao and J. Xie, Precis. Chem., 2024, 2, 495-517.
- 28 Y. Cao, T. Chen, Q. Yao and J. Xie, Acc. Chem. Res., 2021, 54, 4142-4153.
- 29 Z. Luo, V. Nachammai, B. Zhang, N. Yan, D. T. Leong, D.-E. Jiang and J. Xie, J. Am. Chem. Soc., 2014, 136, 10577-10580.
- 30 Q. Yao, T. Chen, X. Yuan and J. Xie, Acc. Chem. Res., 2018, 51, 1338-1348.
- 31 Y. Yu, Q. Yao, K. Cheng, X. Yuan, Z. Luo and J. Xie, Part. Part. Syst. Charact., 2014, 31, 652-656.
- 32 H. Qian and R. Jin, Chem. Mater., 2011, 23, 2209-2217.
- 33 Y. Yu, X. Chen, Q. Yao, Y. Yu, N. Yan and J. Xie, Chem. Mater., 2013, 25, 946-952.
- 34 Y. Yu, Z. Luo, D. M. Chevrier, D. T. Leong, P. Zhang, D.-E. Jiang and J. Xie, J. Am. Chem. Soc., 2014, 136, 1246-1249.
- 35 M. Zhu, Q. Yao, Z. Liu, B. Zhang, Y. Lin, J. Liu, M. Long and J. Xie, Angew. Chem., Int. Ed., 2022, 61, e202115647.
- 36 Z. Yao, B. Sánchez-Lengeling, N. S. Bobbitt, B. J. Bucior, S. G. H. Kumar, S. P. Collins, T. Burns, T. K. Woo,

- O. K. Farha, R. Q. Snurr and A. Aspuru-Guzik, Nat. Mach. Intell., 2021, 3, 76-86.
- 37 M. H. S. Segler, M. Preuss and M. P. Waller, Nature, 2018, 555, 604-610.
- 38 H. Zhao, W. Chen, H. Huang, Z. Sun, Z. Chen, L. Wu, B. Zhang, F. Lai, Z. Wang, M. L. Adam, C. H. Pang, P. K. Chu, Y. Lu, T. Wu, J. Jiang, Z. Yin and X.-F. Yu, Nat. Synth., 2023, 2, 505-514.
- 39 J.-F. Masson, J. S. Biggins and E. Ringe, Nat. Nanotechnol., 2023, 18, 111-123.
- 40 B. J. Shields, J. Stevens, J. Li, M. Parasram, F. Damani, J. I. M. Alvarado, J. M. Janey, R. P. Adams and A. G. Doyle, Nature, 2021, 590, 89-96.
- 41 C. W. Coley, W. H. Green and K. F. Jensen, Acc. Chem. Res., 2018, 51, 1281-1289.
- 42 A. A. Volk, R. W. Epps, D. T. Yonemoto, B. S. Masters, F. N. Castellano, K. G. Reyes and M. Abolhasani, Nat. Commun., 2023, 14, 1403.
- 43 B. C. Haas, D. Kalyani and M. S. Sigman, Sci. Adv., 2025, 11, eadt3013.
- 44 N. J. Szymanski, B. Rendy, Y. Fei, R. E. Kumar, T. He, D. Milsted, M. J. McDermott, M. Gallant, E. D. Cubuk, A. Merchant, H. Kim, A. Jain, C. J. Bartel, K. Persson, Y. Zeng and G. Ceder, Nature, 2023, 624, 86-91.
- 45 Z. Yang, A. Shi, R. Zhang, Z. Ji, J. Li, J. Lyu, J. Qian, T. Chen, X. Wang, F. You and J. Xie, ACS Nano, 2024, 18, 27138-27166.
- 46 J. Li, T. Chen, K. Lim, L. Chen, S. A. Khan, J. Xie and X. Wang, Adv. Intell. Syst., 2019, 1, 1900029.
- 47 E. Sadeghi, P. Mastracco, A. Gonzàlez-Rosell, S. M. Copp and P. Bogdanov, ACS Nano, 2024, 18, 26997–27008.
- 48 S. Li, N.-N. Li, X.-Y. Dong, S.-Q. Zang and T. C. W. Mak, Chem. Rev., 2024, 124, 7262-7378.
- 49 H. Tao, T. Wu, M. Aldeghi, T. C. Wu, A. Aspuru-Guzik and E. Kumacheva, Nat. Rev. Mater., 2021, 6, 701-716.
- 50 L. Rao, Y. Yuan, X. Shen, G. Yu and X. Chen, Nat. Nanotechnol., 2024, 19, 1769-1781.
- 51 K. Kwak, Q. Tang, M. Kim, D.-E. Jiang and D. Lee, J. Am. Chem. Soc., 2015, 137, 10833-10840.
- 52 M. Zhou, T. Higaki, Y. Li, C. Zeng, Q. Li, M. Y. Sfeir and R. Jin, J. Am. Chem. Soc., 2019, 141, 19754-19764.
- 53 Y. Li, Y. Song, X. Zhang, T. Liu, T. Xu, H. Wang, D.-E. Jiang and R. Jin, J. Am. Chem. Soc., 2022, 144, 12381-12389.
- 54 Y. Zhang, P. Song, T. Chen, X. Liu, T. Chen, Z. Wu, Y. Wang, J. Xie and W. Xu, Proc. Natl. Acad. Sci. U. S. A., 2018, 115, 10588-10593.
- 55 S. Li, A. V. Nagarajan, X. Du, Y. Li, Z. Liu, D. R. Kauffman, G. Mpourmpakis and R. Jin, Angew. Chem., Int. Ed., 2022, 61, e202211771.
- 56 K. Kwak, V. D. Thanthirige, K. Pyo, D. Lee and G. Ramakrishna, J. Phys. Chem. Lett., 2017, 8, 4898-4905.
- 57 C. Zeng, Y. Chen, K. Iida, K. Nobusada, K. Kirschbaum, K. J. Lambright and R. Jin, J. Am. Chem. Soc., 2016, 138, 3950-3953.
- 58 Y. Li, M. Zhou, Y. Song, T. Higaki, H. Wang and R. Jin, Nature, 2021, 594, 380-384.

- 59 W. W. Xu, B. Zhu, X. C. Zeng and Y. Gao, Nat. Commun., 2016, 7, 13574.
- 60 L. Xiong, S. Yang, X. Sun, J. Chai, B. Rao, L. Yi, M. Zhu and Y. Pei, J. Phys. Chem. C, 2018, 122, 14898-14907.
- 61 C. Zeng, Y. Chen, C. Liu, K. Nobusada, N. L. Rosi and R. Jin, Sci. Adv., 2015, 1, e1500425.
- 62 W.-Q. Shi, L. Zeng, R.-L. He, X.-S. Han, Z.-J. Guan, M. Zhou and Q.-M. Wang, Science, 2024, 383, 326-330.
- 63 C. Zeng, C. Liu, Y. Chen, N. L. Rosi and R. Jin, J. Am. Chem. Soc., 2014, 136, 11922-11925.
- 64 A. Das, T. Li, K. Nobusada, C. Zeng, N. L. Rosi and R. Jin, J. Am. Chem. Soc., 2013, 135, 18264-18267.
- 65 S. Yang, J. Chai, Y. Song, J. Fan, T. Chen, S. Wang, H. Yu, X. Li and M. Zhu, J. Am. Chem. Soc., 2017, 139, 5668-5671.
- 66 Y. Chen, C. Liu, Q. Tang, C. Zeng, T. Higaki, A. Das, D.-E. Jiang, N. L. Rosi and R. Jin, J. Am. Chem. Soc., 2016, 138, 1482-1485.
- 67 A. Dass, T. Jones, M. Rambukwella, D. Crasto, K. J. Gagnon, L. Sementa, M. De Vetta, O. Baseggio, E. Aprà, M. Stener and A. Fortunelli, J. Phys. Chem. C, 2016, 120, 6256-6261.
- 68 S. M. Reilly, T. Krick and A. Dass, J. Phys. Chem. C, 2010, 114, 741-745.
- 69 L. Xiong, B. Peng, Z. Ma, P. Wang and Y. Pei, Nanoscale, 2017, 9, 2895-2902.
- 70 Y. Wang, C. G. Gianopoulos, Z. Liu, K. Kirschbaum, D. Alfonso, D. R. Kauffman and R. Jin, JACS Au, 2024, 4, 1928-1934.
- 71 C. Zeng, T. Li, A. Das, N. L. Rosi and R. Jin, J. Am. Chem. Soc., 2013, 135, 10011-10013.
- 72 Z. Ma, P. Wang and Y. Pei, Nanoscale, 2016, 8, 17044-17054.
- 73 W. W. Xu, Y. Li, Y. Gao and X. C. Zeng, *Nanoscale*, 2016, 8, 7396-7401.
- 74 S. Takano, S. Yamazoe, K. Koyasu and T. Tsukuda, J. Am. Chem. Soc., 2015, 137, 7027-7030.
- 75 Z. Ma, P. Wang, G. Zhou, J. Tang, H. Li and Y. Pei, J. Phys. Chem. C, 2016, 120, 13739-13748.
- 76 X. Liu, W. W. Xu, X. Huang, E. Wang, X. Cai, Y. Zhao, J. Li, M. Xiao, C. Zhang, Y. Gao, W. Ding and Y. Zhu, Nat. Commun., 2020, 11, 3349.
- 77 H. Dong, L. Liao, S. Zhuang, C. Yao, J. Chen, S. Tian, M. Zhu, X. Liu, L. Li and Z. Wu, Nanoscale, 2017, 9, 3742-3746.
- 78 M. Walter, J. Akola, O. Lopez-Acevedo, P. D. Jadzinsky, G. Calero, C. J. Ackerson, R. L. Whetten, H. Grönbeck and H. Häkkinen, Proc. Natl. Acad. Sci. U. S. A., 2008, 105,
- 79 C. E. Briant, B. R. C. Theobald, J. W. White, L. K. Bell, D. M. P. Mingos and A. J. Welch, J. Chem. Soc., Chem. Commun., 1981, 201-202.
- 80 Y. Shichibu, Y. Negishi, T. Watanabe, N. K. Chaki, H. Kawaguchi and T. Tsukuda, J. Phys. Chem. C, 2007, **111**, 7845-7847.
- 81 H. Qian, W. T. Eckenhoff, Y. Zhu, T. Pintauer and R. Jin, J. Am. Chem. Soc., 2010, 132, 8280-8281.

- 82 Y. Shichibu and K. Konishi, Small, 2010, 6, 1216-1220.
- 83 Y. Song, F. Fu, J. Zhang, J. Chai, X. Kang, P. Li, S. Li, H. Zhou and M. Zhu, Angew. Chem., Int. Ed., 2015, 54, 8430-8434.
- 84 L. C. McKenzie, T. O. Zaikova and J. E. Hutchison, J. Am. Chem. Soc., 2014, 136, 13426-13435.
- 85 X.-K. Wan, Z.-W. Lin and Q.-M. Wang, J. Am. Chem. Soc., 2012, 134, 14750-14752.
- 86 S. Takano, H. Hirai, T. Nakashima, T. Iwasa, T. Taketsugu and T. Tsukuda, J. Am. Chem. Soc., 2021, 143, 10560-10564.
- 87 K. Kwak, W. Choi, Q. Tang, M. Kim, Y. Lee, D.-E. Jiang and D. Lee, Nat. Commun., 2017, 8, 14723.
- 88 S. Takano and T. Tsukuda, J. Am. Chem. Soc., 2021, 143, 1683-1698.
- 89 H. Hirai, S. Takano, T. Nakashima, T. Iwasa, T. Taketsugu and T. Tsukuda, Angew. Chem., Int. Ed., 2022, 61, e202207290.
- 90 Z. Liu, M. Zhou, L. Luo, Y. Wang, E. Kahng and R. Jin, J. Am. Chem. Soc., 2023, 145, 19969-19981.
- 91 Y. Liu, X. Chai, X. Cai, M. Chen, R. Jin, W. Ding and Y. Zhu, Angew. Chem., Int. Ed., 2018, 57, 9775-9779.
- 92 Z. Wu and R. Jin, Nano Lett., 2010, 10, 2568-2573.
- 93 X.-K. Wan, J.-Q. Wang, Z.-A. Nan and Q.-M. Wang, Sci. Adv., 2017, 3, e1701823.
- 94 C. Liu, T. Li, G. Li, K. Nobusada, C. Zeng, G. Pang, N. L. Rosi and R. Jin, Angew. Chem., Int. Ed., 2015, 54, 9826-9829.
- 95 E. L. Albright, T. I. Levchenko, V. K. Kulkarni, A. I. Sullivan, J. F. DeJesus, S. Malola, S. Takano, M. Nambo, K. Stamplecoskie, H. Häkkinen, T. Tsukuda and C. M. Crudden, J. Am. Chem. Soc., 2024, 146, 5759-5780.
- 96 S.-F. Yuan, C.-Q. Xu, W.-D. Liu, J.-X. Zhang, J. Li and Q.-M. Wang, J. Am. Chem. Soc., 2021, 143, 12261-12267.
- 97 J.-H. Huang, Y. Cui, Z.-Y. Wang and S.-Q. Zang, Acc. Chem. Res., 2025, 58, 1249-1261.
- 98 R. Xiao, J. Jia, R. Wang, Y. Feng and H. Chen, Acc. Chem. Res., 2023, 56, 1539-1552.
- 99 W. Kurashige, M. Yamaguchi, K. Nobusada and Y. Negishi, J. Phys. Chem. Lett., 2012, 3, 2649-2652.
- 100 D. M. Chevrier, L. Raich, C. Rovira, A. Das, Z. Luo, Q. Yao, A. Chatt, J. Xie, R. Jin, J. Akola and P. Zhang, J. Am. Chem. Soc., 2018, 140, 15430-15436.
- 101 X. Yuan, N. Goswami, I. Mathews, Y. Yu and J. Xie, Nano Res., 2015, 8, 3488-3495.
- 102 Y. Chen, C. Zeng, D. R. Kauffman and R. Jin, Nano Lett., 2015, 15, 3603-3609.
- 103 X.-K. Wan, Z.-J. Guan and Q.-M. Wang, Angew. Chem., Int. Ed., 2017, 56, 11494-11497.
- 104 X.-K. Wan, Q. Tang, S.-F. Yuan, D.-E. Jiang and Q.-M. Wang, J. Am. Chem. Soc., 2015, 137, 652-655.
- 105 A. Das, T. Li, G. Li, K. Nobusada, C. Zeng, N. L. Rosi and R. Jin, Nanoscale, 2014, 6, 6458-6462.
- 106 Y. Song, S. Wang, J. Zhang, X. Kang, S. Chen, P. Li, H. Sheng and M. Zhu, J. Am. Chem. Soc., 2014, 136, 2963-2965.
- 107 S. Chen, L. Xiong, S. Wang, Z. Ma, S. Jin, H. Sheng, Y. Pei and M. Zhu, J. Am. Chem. Soc., 2016, 138, 10754-10757.

- 108 J.-J. Li, Z.-J. Guan, Z. Lei, F. Hu and Q.-M. Wang, *Angew. Chem., Int. Ed.*, 2019, **58**, 1083–1087.
- 109 Z. Liu, Y. Li, E. Kahng, S. Xue, X. Du, S. Li and R. Jin, ACS Nano, 2022, 16, 18448–18458.
- 110 M. R. Narouz, S. Takano, P. A. Lummis, T. I. Levchenko, A. Nazemi, S. Kaappa, S. Malola, G. Yousefalizadeh, L. A. Calhoun, K. G. Stamplecoskie, H. Häkkinen, T. Tsukuda and C. M. Crudden, J. Am. Chem. Soc., 2019, 141, 14997–15002.
- 111 Q. Yao, X. Yuan, Y. Yu, Y. Yu, J. Xie and J. Y. Lee, *J. Am. Chem. Soc.*, 2015, 137, 2128–2136.
- 112 K. Pyo, V. D. Thanthirige, S. Y. Yoon, G. Ramakrishna and D. Lee, *Nanoscale*, 2016, 8, 20008–20016.
- 113 Y. Cao, S. Malola, M. F. Matus, T. Chen, Q. Yao, R. Shi, H. Häkkinen and J. Xie, *Chem*, 2021, 7, 2227–2244.
- 114 Y. Zhong, J. Zhang, T. Li, W. Xu, Q. Yao, M. Lu, X. Bai, Z. Wu, J. Xie and Y. Zhang, *Nat. Commun.*, 2023, **14**, 658.
- 115 H. Li, X. Kang and M. Zhu, Acc. Chem. Res., 2024, 57, 3194-3205.
- 116 Q. Yao, L. Liu, S. Malola, M. Ge, H. Xu, Z. Wu, T. Chen, Y. Cao, M. F. Matus, A. Pihlajamäki, Y. Han, H. Häkkinen and J. Xie, *Nat. Chem.*, 2023, **15**, 230–239.
- 117 T. Chen, S. Yang, J. Chai, Y. Song, J. Fan, B. Rao, H. Sheng, H. Yu and M. Zhu, *Sci. Adv.*, 2017, 3, e1700956.
- 118 X. Wang, B. Yin, L. Jiang, C. Yang, Y. Liu, G. Zou, S. Chen and M. Zhu, *Science*, 2023, **381**, 784–790.
- 119 H. Liu, G. Hong, Z. Luo, J. Chen, J. Chang, M. Gong, H. He, J. Yang, X. Yuan, L. Li, X. Mu, J. Wang, W. Mi, J. Luo, J. Xie and X.-D. Zhang, *Adv. Mater.*, 2019, 31, 1901015.
- 120 X. Wang, M. B. Liisberg, G. L. Nolt, X. Fu, C. Cerretani, L. Li, L. A. Johnson, T. Vosch and C. I. Richards, *ACS Nano*, 2023, 17, 12862–12874.
- 121 J. Xie, Y. Zheng and J. Y. Ying, *J. Am. Chem. Soc.*, 2009, **131**, 888–889.
- 122 X. Jiang, B. Du, Y. Huang, M. Yu and J. Zheng, *Bioconjugate Chem.*, 2020, **31**, 1522–1528.
- 123 A. I. Sullivan, E. A. Steele, S. Takano, E. Zeinizade, J. Chen, S. Malola, K. Siddhant, H. Häkkinen, K. G. Stamplecoskie, T. Tsukuda, G. Zheng and C. M. Crudden, *J. Am. Chem. Soc.*, 2025, 147, 4230–4238.
- 124 B. Zhang, M. F. Matus, Q. Yao, X. Song, Z. Wu, W. Hu, H. Häkkinen and J. Xie, J. Am. Chem. Soc., 2025, 147, 6404–6414.
- 125 A. Mathew, G. Natarajan, L. Lehtovaara, H. Häkkinen, R. M. Kumar, V. Subramanian, A. Jaleel and T. Pradeep, *ACS Nano*, 2014, **8**, 139–152.
- 126 H. Deng, K. Huang, L. Xiu, W. Sun, Q. Yao, X. Fang, X. Huang, H. A. A. Noreldeen, H. Peng, J. Xie and W. Chen, *Nat. Commun.*, 2022, **13**, 3381.
- 127 Y. Zhang, S.-R. He, Y. Yang, T.-S. Zhang, Z.-M. Zhu, W. Fei and M.-B. Li, *J. Am. Chem. Soc.*, 2023, **145**, 12164–12172.
- 128 P. N. Gunawardene, J. F. Corrigan and M. S. Workentin, *J. Am. Chem. Soc.*, 2019, **141**, 11781–11785.
- 129 P. N. Gunawardene, J. Martin, J. M. Wong, Z. Ding, J. F. Corrigan and M. S. Workentin, *Angew. Chem., Int. Ed.*, 2022, **61**, e202205194.

- 130 V. Rück, N. K. Mishra, K. K. Sørensen, M. B. Liisberg, A. B. Sloth, C. Cerretani, C. B. Mollerup, A. Kjaer, C. Lou, K. J. Jensen and T. Vosch, J. Am. Chem. Soc., 2023, 145, 16771–16777.
- 131 Z. Liu, H. Tan, B. Li, Z. Hu, D.-E. Jiang, Q. Yao, L. Wang and J. Xie, *Nat. Commun.*, 2023, **14**, 3374.
- 132 F. Strieth-Kalthoff, S. Szymkuć, K. Molga, A. Aspuru-Guzik, F. Glorius and B. A. Grzybowski, J. Am. Chem. Soc., 2024, 146, 11005–11017.
- 133 S. Szymkuć, E. P. Gajewska, T. Klucznik, K. Molga, P. Dittwald, M. Startek, M. Bajczyk and B. A. Grzybowski, *Angew. Chem., Int. Ed.*, 2016, 55, 5904–5937.
- 134 M. Brust, M. Walker, D. Bethell, D. J. Schiffrin and R. Whyman, *J. Chem. Soc., Chem. Commun.*, 1994, 801–802.
- 135 Y. Negishi, K. Nobusada and T. Tsukuda, *J. Am. Chem. Soc.*, 2005, **127**, 5261–5270.
- 136 A. Desireddy, B. E. Conn, J. Guo, B. Yoon, R. N. Barnett,
 B. M. Monahan, K. Kirschbaum, W. P. Griffith, R. L. Whetten,
 U. Landman and T. P. Bigioni, *Nature*, 2013, 501, 399–402.
- 137 R. Jin, H. Qian, Z. Wu, Y. Zhu, M. Zhu, A. Mohanty and N. Garg, *J. Phys. Chem. Lett.*, 2010, **1**, 2903–2910.
- 138 Q. Yao, X. Yuan, V. Fung, Y. Yu, D. T. Leong, D.-E. Jiang and J. Xie, *Nat. Commun.*, 2017, **8**, 927.
- 139 C. Zeng, C. Liu, Y. Pei and R. Jin, *ACS Nano*, 2013, 7, 6138–6145.
- 140 S. Wang, H. Abroshan, C. Liu, T.-Y. Luo, M. Zhu, H. J. Kim, N. L. Rosi and R. Jin, *Nat. Commun.*, 2017, 8, 848.
- 141 Y. Lin, Y. Cao, Q. Yao and J. Xie, *Chem. Sci.*, 2022, **13**, 4598–4607.
- 142 T. Chen, V. Fung, Q. Yao, Z. Luo, D.-E. Jiang and J. Xie, *J. Am. Chem. Soc.*, 2018, **140**, 11370–11377.
- 143 T. G. Schaaff, M. N. Shafigullin, J. T. Khoury, I. Vezmar, R. L. Whetten, W. G. Cullen, P. N. First, C. Gutiérrez-Wing, J. Ascensio and M. J. Jose-Yacamán, *J. Phys. Chem. B*, 1997, 101, 7885–7891.
- 144 V. L. Jimenez, D. G. Georganopoulou, R. J. White, A. S. Harper, A. J. Mills, D. Lee and R. W. Murray, *Langmuir*, 2004, **20**, 6864–6870.
- 145 A. Ghosh, J. Hassinen, P. Pulkkinen, H. Tenhu, R. H. A. Ras and T. Pradeep, *Anal. Chem.*, 2014, **86**, 12185–12190.
- 146 Y. Negishi, S. Hashimoto, A. Ebina, K. Hamada, S. Hossain and T. Kawawaki, *Nanoscale*, 2020, **12**, 8017–8039.
- 147 C. Cerretani, H. Kanazawa, T. Vosch and J. Kondo, *Angew. Chem., Int. Ed.*, 2019, **58**, 17153–17157.
- 148 V. Rück, V. A. Neacsu, M. B. Liisberg, C. B. Mollerup, P. H. Ju, T. Vosch, J. Kondo and C. Cerretani, *Adv. Opt. Mater.*, 2024, **12**, 2301928.
- 149 C.-Y. Liu, S.-F. Yuan, S. Wang, Z.-J. Guan, D.-E. Jiang and Q.-M. Wang, *Nat. Commun.*, 2022, 13, 2082.
- 150 Y. Shichibu, Y. Negishi, H. Tsunoyama, M. Kanehara, T. Teranishi and T. Tsukuda, *Small*, 2007, 3, 835–839.
- 151 H. Qian, M. Zhu, U. N. Andersen and R. Jin, *J. Phys. Chem. A*, 2009, **113**, 4281–4284.
- 152 W. Du, S. Jin, L. Xiong, M. Chen, J. Zhang, X. Zou, Y. Pei, S. Wang and M. Zhu, J. Am. Chem. Soc., 2017, 139, 1618–1624.

- 153 T. A. Dreier and C. J. Ackerson, Angew. Chem., Int. Ed., 2015, 54, 9249-9252.
- 154 Y. Cao, T. Liu, T. Chen, B. Zhang, D.-E. Jiang and J. Xie, Nat. Commun., 2021, 12, 3212.
- 155 Y. Cao, V. Fung, Q. Yao, T. Chen, S. Zang, D.-E. Jiang and J. Xie, Nat. Commun., 2020, 11, 5498.
- 156 C. Yan, Y. Li, E. Wang and W. W. Xu, J. Phys. Chem. Lett., 2023, 14, 7632-7637.
- 157 M. A. Tofanelli, K. Salorinne, T. W. Ni, S. Malola, B. Newell, B. Phillips, H. Häkkinen and C. J. Ackerson, Chem. Sci., 2016, 7, 1882-1890.
- 158 T. Dainese, S. Antonello, S. Bogialli, W. Fei, A. Venzo and F. Maran, ACS Nano, 2018, 12, 7057-7066.
- 159 S. Yang, S. Chen, L. Xiong, C. Liu, H. Yu, S. Wang, N. L. Rosi, Y. Pei and M. Zhu, J. Am. Chem. Soc., 2018, 140, 10988-10994.
- 160 B. Zhang, C. Chen, W. Chuang, S. Chen and P. Yang, J. Am. Chem. Soc., 2020, 142, 11514-11520.
- 161 X. S. Han, X. Luan, H. F. Su, J. J. Li, S. F. Yuan, Z. Lei, Y. Pei and Q. M. Wang, Angew. Chem., Int. Ed., 2020, 59, 2309-2312.
- 162 A. Das, C. Liu, H. Y. Byun, K. Nobusada, S. Zhao, N. Rosi and R. Jin, Angew. Chem., Int. Ed., 2015, 54, 3140-3144.
- 163 S. Chen, S. Wang, J. Zhong, Y. Song, J. Zhang, H. Sheng, Y. Pei and M. Zhu, Angew. Chem., Int. Ed., 2015, 54, 3145-3149.
- 164 A. Dass, T. C. Jones, S. Theivendran, L. Sementa and A. Fortunelli, J. Phys. Chem. C, 2017, 121, 14914-14919.
- 165 M. P. Maman, A. S. Nair, H. Cheraparambil, B. Pathak and S. Mandal, J. Phys. Chem. Lett., 2020, 11, 1781-1788.
- 166 Q. Yao, V. Fung, C. Sun, S. Huang, T. Chen, D.-E. Jiang, J. Y. Lee and J. Xie, Nat. Commun., 2018, 9, 1979.
- 167 M. S. Bootharaju, C. P. Joshi, M. R. Parida, O. F. Mohammed and O. M. Bakr, Angew. Chem., Int. Ed., 2016, 55, 922-926.
- 168 Y. Negishi, T. Iwai and M. Ide, Chem. Commun., 2010, 46, 4713-4715.
- 169 Y. Negishi, K. Munakata, W. Ohgake and K. Nobusada, J. Phys. Chem. Lett., 2012, 3, 2209-2214.
- 170 L. V. Nair, S. Hossain, S. Takagi, Y. Imai, G. Hu, S. Wakayama, B. Kumar, W. Kurashige, D.-E. Jiang and Y. Negishi, Nanoscale, 2018, 10, 18969-18979.
- 171 Q. Li, K. J. Lambright, M. G. Taylor, K. Kirschbaum, T.-Y. Luo, J. Zhao, G. Mpourmpakis, S. Mokashi-Punekar, N. L. Rosi and R. Jin, J. Am. Chem. Soc., 2017, 139, 17779-17782.
- 172 L. Liao, S. Zhou, Y. Dai, L. Liu, C. Yao, C. Fu, J. Yang and Z. Wu, J. Am. Chem. Soc., 2015, 137, 9511-9514.
- 173 Z.-H. Gao, K. Wei, T. Wu, J. Dong, D.-E. Jiang, S. Sun and L.-S. Wang, J. Am. Chem. Soc., 2022, 144, 5258-5262.
- 174 T.-H. Chiu, J.-H. Liao, F. Gam, Y.-Y. Wu, X. Wang, S. Kahlal, J.-Y. Saillard and C. W. Liu, J. Am. Chem. Soc., 2022, 144, 10599-10607.
- 175 W. Fei, S. Antonello, T. Dainese, A. Dolmella, M. Lahtinen, K. Rissanen, A. Venzo and F. Maran, J. Am. Chem. Soc., 2019, 141, 16033-16045.

- 176 S. Yamazoe, W. Kurashige, K. Nobusada, Y. Negishi and T. Tsukuda, J. Phys. Chem. C, 2014, 118, 25284-25290.
- 177 H. Qian, D.-e Jiang, G. Li, C. Gayathri, A. Das, R. R. Gil and R. Jin, J. Am. Chem. Soc., 2012, 134, 16159-16162.
- 178 M. Sera, S. Hossain, S. Yoshikawa, K. Takemae, A. Ikeda, T. Tanaka, T. Kosaka, Y. Niihori, T. Kawawaki and Y. Negishi, J. Am. Chem. Soc., 2024, 146, 29684-29693.
- 179 C. Kumara and A. Dass, Nanoscale, 2012, 4, 4084-4086.
- 180 Y. Niihori, Y. Koyama, S. Watanabe, S. Hashimoto, S. Hossain, L. V. Nair, B. Kumar, W. Kurashige and Y. Negishi, J. Phys. Chem. Lett., 2018, 9, 4930-4934.
- 181 Y. Niihori, S. Hashimoto, Y. Koyama, S. Hossain, W. Kurashige and Y. Negishi, J. Phys. Chem. C, 2019, 123, 13324-13329.
- 182 C. Yao, Y.-j Lin, J. Yuan, L. Liao, M. Zhu, L.-h Weng, J. Yang and Z. Wu, J. Am. Chem. Soc., 2015, 137, 15350-15353.
- 183 Z. Gan, N. Xia and Z. Wu, Acc. Chem. Res., 2018, 51, 2774-2783.
- 184 K. Zheng, V. Fung, X. Yuan, D.-E. Jiang and J. Xie, J. Am. Chem. Soc., 2019, 141, 18977-18983.
- 185 Q. Yao, Y. Feng, V. Fung, Y. Yu, D.-E. Jiang, J. Yang and J. Xie, Nat. Commun., 2017, 8, 1555.
- 186 Q. Li, T.-Y. Luo, M. G. Taylor, S. Wang, X. Zhu, Y. Song, G. Mpourmpakis, N. L. Rosi and R. Jin, Sci. Adv., 2017, 3, e1603193.
- 187 Q. Li, S. Wang, K. Kirschbaum, K. J. Lambright, A. Das and R. Jin, Chem. Commun., 2016, 52, 5194-5197.
- 188 S. Wang, Y. Song, S. Jin, X. Liu, J. Zhang, Y. Pei, X. Meng, M. Chen, P. Li and M. Zhu, J. Am. Chem. Soc., 2015, 137, 4018-4021.
- 189 M. Zhu, P. Wang, N. Yan, X. Chai, L. He, Y. Zhao, N. Xia, C. Yao, J. Li, H. Deng, Y. Zhu, Y. Pei and Z. Wu, Angew. Chem., Int. Ed., 2018, 57, 4500-4504.
- 190 K. R. Krishnadas, A. Baksi, A. Ghosh, G. Natarajan and T. Pradeep, Nat. Commun., 2016, 7, 13447.
- 191 M. Neumaier, A. Baksi, P. Weis, E. K. Schneider, P. Chakraborty, H. Hahn, T. Pradeep and M. M. Kappes, J. Am. Chem. Soc., 2021, 143, 6969-6980.
- 192 X. Yuan, B. Zhang, Z. Luo, Q. Yao, D. T. Leong, N. Yan and J. Xie, Angew. Chem., Int. Ed., 2014, 53, 4623-4627.
- 193 J.-H. Huang, Z.-Y. Wang, S.-Q. Zang and T. C. W. Mak, ACS Cent. Sci., 2020, 6, 1971-1976.
- 194 H. Shen, Z. Xu, M. S. A. Hazer, Q. Wu, J. Peng, R. Qin, S. Malola, B. K. Teo, H. Häkkinen and N. Zheng, Angew. Chem., Int. Ed., 2021, 60, 3752-3758.
- 195 T. W. Ni, M. A. Tofanelli, B. D. Phillips and C. J. Ackerson, Inorg. Chem., 2014, 53, 6500-6502.
- 196 Y. Niihori, Y. Kikuchi, A. Kato, M. Matsuzaki and Y. Negishi, ACS Nano, 2015, 9, 9347-9356.
- 197 Z. Lei, J.-J. Li, Z.-A. Nan, Z.-G. Jiang and Q.-M. Wang, Angew. Chem., Int. Ed., 2021, 60, 14415-14419.
- 198 B. M. Barngrover and C. M. Aikens, J. Am. Chem. Soc., 2012, 134, 12590-12595.
- 199 J. S. Kim, H. Chang, S. Kang, S. Cha, H. Cho, S. J. Kwak, N. Park, Y. Kim, D. Kang, C. K. Song, J. Kwag, J.-S. Hahn, W. B. Lee, T. Hyeon and J. Park, Nat. Commun., 2023, 14, 3201.

- 200 Y. G. Srinivasulu, N. Goswami, Q. Yao and J. Xie, J. Phys. Chem. C, 2021, 125, 4066–4076.
- 201 T. Chen, Q. Yao, X. Yuan, R. R. Nasaruddin and J. Xie, J. Phys. Chem. C, 2017, 121, 10743–10751.
- 202 M. Zhu, E. Lanni, N. Garg, M. E. Bier and R. Jin, J. Am. Chem. Soc., 2008, 130, 1138-1139.
- 203 X. Meng, Z. Liu, M. Zhu and R. Jin, *Nanoscale Res. Lett.*, 2012, 7, 277.
- 204 O. Cybulski, C. Quintana, M. Siek and B. A. Grzybowski, *Small*, 2024, **20**, 2400306.
- 205 T. Chen, Z. Luo, Q. Yao, A. X. H. Yeo and J. Xie, *Chem. Commun.*, 2016, 52, 9522–9525.
- 206 L. Liao, S. Zhuang, C. Yao, N. Yan, J. Chen, C. Wang, N. Xia, X. Liu, M.-B. Li, L. Li, X. Bao and Z. Wu, *J. Am. Chem. Soc.*, 2016, 138, 10425–10428.
- 207 L. Liao, S. Zhuang, P. Wang, Y. Xu, N. Yan, H. Dong, C. Wang, Y. Zhao, N. Xia, J. Li, H. Deng, Y. Pei, S.-K. Tian and Z. Wu, *Angew. Chem.*, *Int. Ed.*, 2017, 56, 12644–12648.
- 208 X. Lin, J. Tang, C. Zhu, L. Wang, Y. Yang, R. A. Wu, H. Fan, C. Liu and J. Huang, *Chem. Sci.*, 2023, 14, 994–1002.
- 209 Q. Yao, Y. Yu, X. Yuan, Y. Yu, J. Xie and J. Y. Lee, *Small*, 2013, **9**, 2696–2701.
- 210 Z. Wu, J. Suhan and R. Jin, J. Mater. Chem., 2009, 19, 622–626.
- 211 T. Chen, Q. Yao, Y. Cao and J. Xie, *Cell Rep. Phys. Sci.*, 2020, **1**, 100206.
- 212 Z. Wu, M. A. MacDonald, J. Chen, P. Zhang and R. Jin, *J. Am. Chem. Soc.*, 2011, **133**, 9670–9673.
- 213 F. Bertorelle, I. Russier-Antoine, C. Comby-Zerbino, F. Chirot, P. Dugourd, P.-F. Brevet and R. Antoine, *ACS Omega*, 2018, 3, 15635–15642.
- 214 A. Ghosh, T. Udayabhaskararao and T. Pradeep, *J. Phys. Chem. Lett.*, 2012, **3**, 1997–2002.
- 215 S. Zhuang, L. Liao, M.-B. Li, C. Yao, Y. Zhao, H. Dong, J. Li, H. Deng, L. Li and Z. Wu, *Nanoscale*, 2017, 9, 14809–14813.
- 216 S. Zhuang, L. Liao, Y. Zhao, J. Yuan, C. Yao, X. Liu, J. Li, H. Deng, J. Yang and Z. Wu, *Chem. Sci.*, 2018, 9, 2437–2442.
- 217 S. Zhuang, L. Liao, J. Yuan, N. Xia, Y. Zhao, C. Wang, Z. Gan, N. Yan, L. He, J. Li, H. Deng, Z. Guan, J. Yang and Z. Wu, *Angew. Chem., Int. Ed.*, 2019, 58, 4510–4514.
- 218 C. J. Taylor, A. Pomberger, K. C. Felton, R. Grainger, M. Barecka, T. W. Chamberlain, R. A. Bourne, C. N. Johnson and A. A. Lapkin, *Chem. Rev.*, 2023, 123, 3089–3126.
- 219 A. M. K. Nambiar, C. P. Breen, T. Hart, T. Kulesza, T. F. Jamison and K. F. Jensen, *ACS Cent. Sci.*, 2022, 8, 825–836.
- 220 N. I. Rinehart, R. K. Saunthwal, J. Wellauer, A. F. Zahrt, L. Schlemper, A. S. Shved, R. Bigler, S. Fantasia and S. E. Denmark, *Science*, 2023, 381, 965–972.
- 221 G. Panapitiya, G. Avendaño-Franco, P. Ren, X. Wen, Y. Li and J. P. Lewis, *J. Am. Chem. Soc.*, 2018, **140**, 17508–17514.
- 222 S. Wang, Z. Wu, S. Dai and D.-E. Jiang, *Angew. Chem., Int. Ed.*, 2021, **60**, 12289–12292.
- 223 S. Wang, T. Liu and D.-E. Jiang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 53468–53474.

- 224 A. Pihlajamäki, M. F. Matus, S. Malola and H. Häkkinen, Adv. Mater., 2024, 36, 2407046.
- 225 A. Pihlajamäki, J. Hämäläinen, J. Linja, P. Nieminen, S. Malola, T. Kärkkäinen and H. Häkkinen, *J. Phys. Chem. A*, 2020, **124**, 4827–4836.
- 226 S. Malola, P. Nieminen, A. Pihlajamäki, J. Hämäläinen, T. Kärkkäinen and H. Häkkinen, *Nat. Commun.*, 2019, 10, 3973.
- 227 F. Zhai, Y. Guan, Y. Li, S. Chen and R. He, *ACS Appl. Nano Mater.*, 2022, 5, 9615–9624.
- 228 L. Tang, L. Wang, B. Wang, Y. Pei and S. Wang, *Chem. Eur. J.*, 2024, **30**, e202302602.
- 229 T. Chen, J. Li, P. Cai, Q. Yao, Z. Ren, Y. Zhu, S. Khan, J. Xie and X. Wang, *Nano Res.*, 2023, 16, 4188–4196.
- 230 S. M. Copp, A. Gorovits, S. M. Swasey, S. Gudibandi, P. Bogdanov and E. G. Gwinn, ACS Nano, 2018, 12, 8240–8247.
- 231 P. Mastracco, A. Gonzàlez-Rosell, J. Evans, P. Bogdanov and S. M. Copp, *ACS Nano*, 2022, **16**, 16322–16331.
- 232 K. Jin, W. Wang, G. Qi, X. Peng, H. Gao, H. Zhu, X. He, H. Zou, L. Yang, J. Yuan, L. Zhang, H. Chen and X. Qu, *Nanoscale*, 2023, 15, 15358–15367.
- 233 T.-R. Tero, S. Malola, B. Koncz, E. Pohjolainen, S. Lautala, S. Mustalahti, P. Permi, G. Groenhof, M. Pettersson and H. Häkkinen, ACS Nano, 2017, 11, 11872–11879.
- 234 A. Pihlajamäki, S. Malola, T. Kärkkäinen and H. Häkkinen, *J. Phys. Chem. C*, 2023, **127**, 14211–14221.
- 235 J. A. G. Torres, S. H. Lau, P. Anchuri, J. M. Stevens, J. E. Tabora, J. Li, A. Borovika, R. P. Adams and A. G. Doyle, *J. Am. Chem. Soc.*, 2022, 144, 19999–20007.
- 236 M.-H. Du, Y. Dai, L.-P. Jiang, Y.-M. Su, M.-Q. Qi, C. Wang, L.-S. Long, L.-S. Zheng and X.-J. Kong, *J. Am. Chem. Soc.*, 2023, 145, 23188–23195.
- 237 C. Cai, S. Wang, Y. Xu, W. Zhang, K. Tang, Q. Ouyang, L. Lai and J. Pei, *J. Med. Chem.*, 2020, **63**, 8683–8694.
- 238 C. Jiang, H. He, H. Guo, X. Zhang, Q. Han, Y. Weng, X. Fu, Y. Zhu, N. Yan, X. Tu and Y. Sun, *Nat. Commun.*, 2024, 15, 6301.
- 239 E. Shim, J. A. Kammeraad, Z. Xu, A. Tewari, T. Cernak and P. M. Zimmerman, *Chem. Sci.*, 2022, **13**, 6655–6668.
- 240 X. Xue, H. Sun, M. Yang, X. Liu, H.-Y. Hu, Y. Deng and X. Wang, *Anal. Chem.*, 2023, **95**, 13733–13745.
- Z. Yang, Y.-M. Zhao, X. Wang, X. Liu, X. Zhang, Y. Li, Q. Lv,C. Y.-C. Chen and L. Shen, *Nat. Commun.*, 2024, 15, 8148.
- 242 L.-Y. Chen and Y.-P. Li, Nat. Commun., 2025, 16, 3262.
- 243 X. Zhong, B. Gallagher, S. Liu, B. Kailkhura, A. Hiszpanski and T. Y.-J. Han, *npj Comput. Mater.*, 2022, **8**, 204.
- 244 B. B. Mendes, Z. Zhang, J. Conniot, D. P. Sousa, J. M. J. M. Ravasco, L. A. Onweller, A. Lorenc, T. Rodrigues, D. Reker and J. Conde, *Nat. Nanotechnol.*, 2024, 19, 867–878.
- 245 M. Suvarna, T. Zou, S. H. Chong, Y. Ge, A. J. Martín and J. Pérez-Ramírez, *Nat. Commun.*, 2024, **15**, 5844.
- 246 A. Rao and M. Grzelczak, *Chem. Mater.*, 2024, 36, 2577–2587.
- 247 T.-S. Vu, M.-Q. Ha, D.-N. Nguyen, V.-C. Nguyen, Y. Abe, T. Tran, H. Tran, H. Kino, T. Miyake, K. Tsuda and H.-C. Dam, *npj Comput. Mater.*, 2023, **9**, 215.

- 248 G. Wu, H. Zhou, J. Zhang, Z.-Y. Tian, X. Liu, S. Wang, C. W. Coley and H. Lu, Nat. Synth., 2023, 2, 515-526.
- 249 M. D. Wilkinson, M. Dumontier, I. J. Aalbersberg, G. Appleton, M. Axton, A. Baak, N. Blomberg, J.-W. Boiten, L. B. da Silva Santos, P. E. Bourne, J. Bouwman, A. J. Brookes, T. Clark, M. Crosas, I. Dillo, O. Dumon, S. Edmunds, C. T. Evelo, R. Finkers, A. Gonzalez-Beltran, A. J. G. Gray, P. Groth, C. Goble, J. S. Grethe, J. Heringa, P. A. C. Hoen, R. Hooft, T. Kuhn, R. Kok, J. Kok, S. J. Lusher, M. E. Martone, A. Mons, A. L. Packer, B. Persson, P. Rocca-Serra, M. Roos, R. van Schaik, S.-A. Sansone, E. Schultes, T. Sengstag, T. Slater, G. Strawn, M. A. Swertz, M. Thompson, J. van der Lei, E. van Mulligen, J. Velterop, A. Waagmeester, P. Wittenburg, K. Wolstencroft, J. Zhao and B. Mons, Sci. Data, 2016, 3, 160018.
- 250 H. Wang, T. Fu, Y. Du, W. Gao, K. Huang, Z. Liu, P. Chandak, S. Liu, P. Van Katwyk, A. Deac, A. Anandkumar, K. Bergen, C. P. Gomes, S. Ho, P. Kohli, J. Lasenby, J. Leskovec, T.-Y. Liu, A. Manrai, D. Marks, B. Ramsundar, L. Song, J. Sun, J. Tang, P. Veličković, M. Welling, L. Zhang, C. W. Coley, Y. Bengio and M. Zitnik, Nature, 2023, 620, 47-60.
- 251 C. W. Coley, N. S. Eyke and K. F. Jensen, Angew. Chem., Int. Ed., 2020, 59, 23414-23436.
- 252 Z. Zheng, O. Zhang, C. Borgs, J. T. Chayes and O. M. Yaghi, J. Am. Chem. Soc., 2023, 145, 18048-18062.

- 253 J. Dagdelen, A. Dunn, S. Lee, N. Walker, A. S. Rosen, G. Ceder, K. A. Persson and A. Jain, Nat. Commun., 2024, **15**, 1418.
- 254 D. A. Boiko, R. MacKnight, B. Kline and G. Gomes, Nature, 2023, 624, 570-578.
- 255 B. Sanchez-Lengeling and A. Aspuru-Guzik, Science, 2018, **361**. 360-365.
- 256 A. Ilnicka and G. Schneider, Nat. Comput. Sci., 2023, 3, 922-933.
- 257 C. Zeni, R. Pinsler, D. Zügner, A. Fowler, M. Horton, X. Fu, Z. Wang, A. Shysheya, J. Crabbé, S. Ueda, R. Sordillo, L. Sun, J. Smith, B. Nguyen, H. Schulz, S. Lewis, C.-W. Huang, Z. Lu, Y. Zhou, H. Yang, H. Hao, J. Li, C. Yang, W. Li, R. Tomioka and T. Xie, Nature, 2025, 639, 624-632.
- 258 F. Therrien, E. H. Sargent and O. Voznyy, Nat. Commun., 2025, 16, 4301.
- 259 Y. Zheng, J. C. Schupp, T. Adams, G. Clair, A. Justet, F. Ahangari, X. Yan, P. Hansen, M. Carlon, E. Cortesi, M. Vermant, R. Vos, L. J. De Sadeleer, I. O. Rosas, R. Pineda, J. Sembrat, M. Königshoff, J. E. McDonough, B. M. Vanaudenaerde, W. A. Wuyts, N. Kaminski and J. Ding, Nat. Biomed. Eng., 2025, DOI: 10.1038/s41551-025-01423-7.
- 260 T. Rahman, A. Tahmid, S. E. Arman, T. Ahmed, Z. T. Rakhy, H. Das, M. Rahman, A. K. Azad, M. Wahadoszamen and A. Habib, Nanoscale Adv., 2025, 7, 634-642.