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# Chemical models to map the transformation from precursors to semiconductor materials at non-equilibrium conditions†

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This work proposes the construction of chemical models based on the Gibbs composition triangle, which provides support for the proper interpretation of semiconductor synthesis under non-equilibrium processing, considering the pertinent variables of the system. It demonstrates how chemical models are constructed using experimental findings and theoretical insights and by incorporating data available in the literature. Then, an illustrative example is used to validate the construction, interpretation and application of a chemical model for obtaining PbTe *via* non-equilibrium process. This approach can be directly applied to forecast the formation of IV–VI and II–VI binary semiconductors, as well as the formation of ternary semiconductor solid solutions. However, it is exemplified—in this work—*via* the mechanochemical synthesis of PbTe. This work aims to construct a chemical model that maps the transformation from precursors to semiconductor material through the high-energy milling process.

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## Introduction

Over the last decade, the research, development, and production of materials and alloys have intensified, significantly improving our daily lives. Most of the alloys obtained are based on phase diagrams because they can assist in designing experimental routes to obtain certain phases with intriguing properties. Phase diagrams also allow for determining the behavior—thermodynamic stability—of the different phases involved in a system. In other words, they can help us understand how temperature, pressure, and composition affect the microstructure in specific phases. For this reason, phase diagrams are indispensable for the development of alloys, which are widely applied in various fields of science and technology.

The phase diagrams are considered roadmaps that denote the relationships between the phases present, temperature, pressure, and composition that occur under specific processing conditions. If the phase diagram of an alloy system is known, then the preparation of the alloy with desired properties can be controlled. In practice, the use of phase diagrams avoids the experimental trial-and-error method, which costs considerable time, effort, and resources. Additionally, phase diagrams have applications in manufacturing, such as thermomechanical processes and heat treatments, among others.

Although phase diagrams are currently used to obtain alloys at or near equilibrium conditions,<sup>1,2</sup> significant advancements have been made in the field of phase diagrams nowadays. For instance, theoretical modelling of phase diagrams of nanomaterials,<sup>3,4</sup> nanoalloy phase diagrams,<sup>5</sup> size-dependent phase diagrams,<sup>6,7</sup> among others, provides new opportunities not only in understanding new systems but also in materials design.<sup>4,8,9</sup> Unambiguously, the results obtained using computer simulations and/or experimental findings are necessary for calculating and constructing more reliable phase diagrams of alloy systems.<sup>10–13</sup>

Until now, most experimental routes to obtain materials and systems at equilibrium have focused on phase diagrams, but alternatives to non-equilibrium processing conditions are lacking. It is essential to note that, over the last few decades, non-equilibrium processes have taken advantage of the versatility of precursor and reaction parameters (such as processing time, room temperature, atmospheric conditions, and multi-step strategies) to produce unique nanomaterials for the fabrication of increasingly impressive nanodevices.<sup>14–17</sup> Unfortunately, there is insufficient literature available for the interpretation and application of graphical methods, such as phase diagrams, in materials production *via* non-equilibrium processes. It is well known that reported phase diagrams, thermodynamic databases, and specialized scientific software are frequently used to design experimental routes for obtaining alloys with specific phases. Lamentably, that kind of data, including graphical methods (road maps), in systems not at equilibrium are scarce. For that reason, to construct diagrams for alloy production *via* non-equilibrium processes remains a challenge in materials science and related fields.

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This work aims to construct chemical models (roadmaps) that map the transformation from precursors to semiconductor materials through the high-energy milling (HEM) process, which is currently achieved *via* synthesis under non-equilibrium conditions. Such chemical models are usable and compatible, in terms of comprehension, with current processing practices. Additionally, this proposal opens up new opportunities for visualizing the chemical pathway during the synthesis of semiconductors under non-equilibrium conditions.

## Preliminary considerations

When non-equilibrium conditions are considered during the processing of materials, the data reported in phase diagrams may be inconsistent with experimental findings, *i.e.*, phase and microstructure changes are not reasonably explained. Therefore, phase diagrams cannot assist in forecasting phase relations, compositional changes and structures in systems under far-from-equilibrium conditions. Thus, non-equilibrium processing of materials often differs significantly from the data available in phase diagrams. However, the transformation from precursors to semiconductor materials can be traced by considering three significant reaction stages during the mechanochemical synthesis of IV–VI and II–VI binary semiconductors, including the formation of ternary semiconductor solid solutions; namely, (1) oxidation, including reoxidation of precursors, (2) chemical interaction between both metal and chalcogen suboxides to give way to stoichiometric complex oxides and (3) chemical reduction of that complex oxides to obtain semiconductor materials. Such a series of both solid–solid and gas–solid reactions are generated in the course of the HEM process, and the reaction products are obtained at specific prefixed milling time by considering local equilibria.<sup>18</sup>

## Experimental considerations

As is known, during milling, sampling is performed by taking powder-milled mixtures from the vial at several prefixed periods of processing time to characterize the as-milled sample to permit researchers gaining an in-depth understanding of the relationship between milling time and coexisting phases at all length scales, including bulk and surface one, see ESI† for further details on sample preparation. For phase non-equilibrium studies, standard bulk experimental techniques are required, *e.g.*, X-ray diffraction (XRD), secondary ion mass spectrometry (SIMS), electron-probe microanalysis (EPMA), and scanning electron microscopy (SEM), among others. Additionally, the former must be supplemented with accurate surface experimental measurements, *e.g.*, high-resolution transmission electron microscopy (HRTEM), scanning transmission electron microscopy (STEM), and X-ray photoelectron spectroscopy (XPS).

As reported in the literature, the experimental findings obtained through bulk and surface measurements have, in fact, begun to serve as a guide at each prefixed processing time.<sup>18–22</sup> Under composition, pressure, and temperature (c–P–T) vial

conditions, it enables an analysis of the significant relationship between interatomic chemical interactions and phase transitions that occur in materials with varying milling times. Then, when the structure and composition of all phases in a system are known as a function of processing time, considering local equilibrium, a preliminary chemical model can be proposed to forecast the formation of a material. A detailed description of the relevant data of each phase in this case study can be found in the ESI.†

## Theoretical considerations

Emphasis is placed on describing aspects that involve chemical transformations during the HEM process through the graphical representation approach, *i.e.*, the chemical model. To demonstrate how local and overall chemical kinetics contribute to the global transformations from precursors to the final product, a framework of experimental and theoretical reactions is proposed as a function of milling time in accordance with the thermodynamic data available in the literature.<sup>19,23,24</sup> Lastly, a thorough explanation of the phase transition chemical pathway linked to the non-equilibrium process, from precursors to product, can be proposed. Consequently, the chemical model involves an interpretation whose forecasts are verified through experimental findings, theoretical data, and even analogous and ideal phases that have not been documented to date. It is worth mentioning that possible gas phases were considered in terms of Gibbs free energy.<sup>19,23,24</sup>

## Chemical model construction

Based on experimental findings and theoretical considerations, it is established that the chemical model encompasses three primary stages associated with a series of phases, which are a function of both milling atmosphere (oxygen potential) and processing time, provided that precursors are poured under atmospheric conditions.<sup>18–22</sup> In addition to this, the formation of semiconductors *via* a non-equilibrium process results from an interplay between physico-metallurgical and mechanochemical stimuli.<sup>23</sup> The first kind of stimulus tends to favor agglomeration of particles because of the Van der Waals forces, dis-agglomeration and re-agglomeration of tiny particles, plastic deformation, fracture and mechanical kneading as well. The second kind is related to the oxidation and reoxidation of precursors, intermediate reactions, and transitions where crystalline, amorphous, and/or non-stoichiometric phases are formed at different scales. The interaction between physico-metallurgical and mechano-chemical stimuli has resulted in the production of semiconductors with increasingly more elaborated shapes and structures when specific process control agents (PCA) are used.<sup>25–28</sup>

A chemical model is constructed using slice cuts, which are a function of processing time, considering oxygen potential while taking into account other parameters (*e.g.*, type of mill, milling media, vial composition, PCA addition, *etc.*). Such slice cuts of the chemical model reveal the phases present associated with a specific processing time, which, in turn, allows us to infer



the chemical reactions undergone by the precursors (Fig. 1). Therefore, to avoid the presence of impurities during the semiconductors processing, an inert experimental setup is taken into consideration, see the ESI† for further details on sample preparation.<sup>18–25,27,28</sup>

## Example of construction and discussion

The proposal of graphical methods to describe materials formation adequately at non-equilibrium processing conditions is lacking, hindering the design of experimental routes to obtain specific materials with desired properties. Currently, there is no proposal to address this topic, making it difficult for researchers to visualize a possible path related to non-equilibrium processes. Therefore, to establish how, during milling, a system is driven from precursors to a mixture of phases, including oxides, hence to the desired semiconductor, a chemical model is constructed.

### General features for the construction of the chemical model

In this work, it is shown that an appropriate chemical model for the mechanochemical synthesis of PbTe can be constructed using both bulk and surface characterization, supplemented with theoretical data reported in the literature. This approach

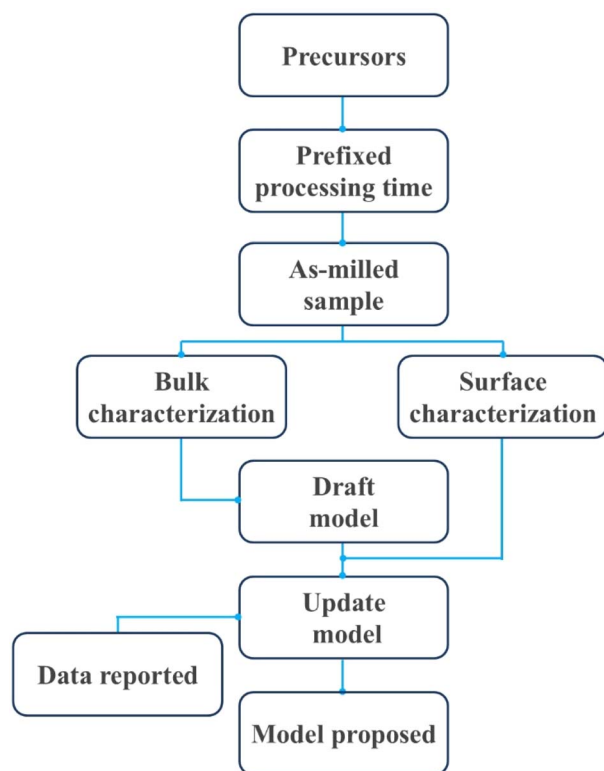


Fig. 1 Schematic summarizing the chemical model construction used to obtain a description for the mechanochemical synthesis of semiconductors.

considers milling time and thermodynamic data available in the literature, as shown in Fig. 2.

At this point, starting from the very basics, it is important to keep in mind that although the Pb–Te system is the main, oxygen must be taken into account due to the milling is carried out under an air atmosphere and/or because it is present in the precursors (*e.g.*, lead oxide). As a consequence, the Pb–Te–O system is considered the central system in the Gibbs composition triangle, as shown in Fig. 2a. As shown in this figure, it can be inferred that apexes represent pure substances, *i.e.*, Pb, Te, and O. Moreover, three binary systems are given in the Gibbs composition triangle as Pb–O, Pb–Te and Te–O. Then, after

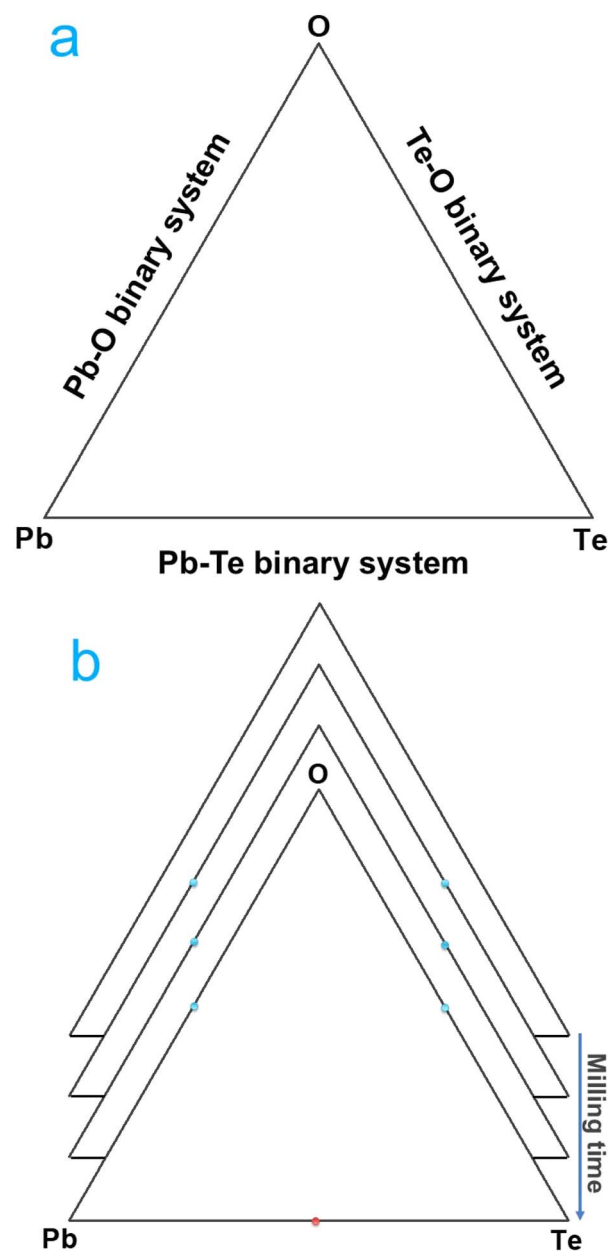


Fig. 2 (a) Schematic initial chemical model construction. (b) Schematic overlay of the experimental findings traced as a function of the processing time.



each prefixed milling time, the experimental stoichiometric phases detected can be consecutively located in the Pb–Te–O system and/or along their respective binaries (see Fig. 2b). A detailed description of the relevant data for each phase can be found in the ESI.†

### Location of phases in the chemical model

The first stage, the initial chemical condition, is denoted by the starting point  $S_i$ , which depends on the chemical composition of the precursors. For instance, if PbO and Te are the precursors,  $S_i$  will be denoted by I. On the contrary, if Pb and Te are the precursors,  $S_i$  will be denoted by II, as shown in Fig. 3a.

As shown in Fig. 3b, due to milling being carried out under an air atmosphere, oxidation or reoxidation of the precursors is expected. Consequently, phases are arranged in the Pb–Te–O system according to the usual chemical formulae of the

substances. Furthermore, as can be noted in the Pb–O and Te–O binary systems, Pb- and Te-oxides are located as molecular stoichiometric species, which depend on the oxygen content. It is worth noting that extreme Te-oxidation did not occur during the HEM process.

As the milling time progresses, Pb- and Te-oxides react with each other to form a complex oxide with the maximum quantity of oxygen. In this case, it is represented by  $\text{PbTeO}_3$ , which is stoichiometrically balanced (see Fig. 4a). As can be noted,

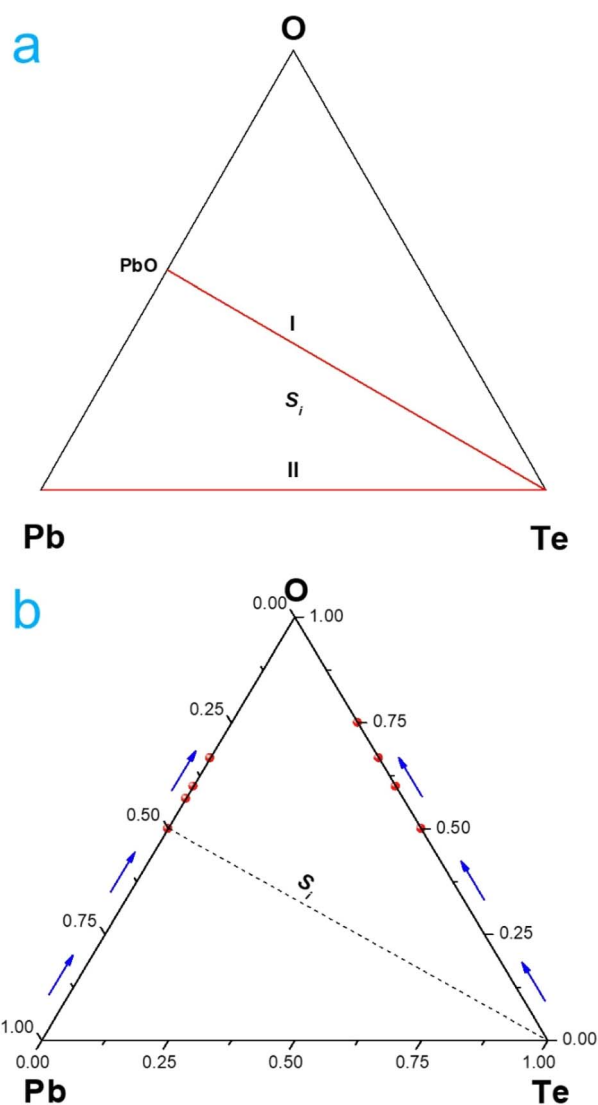


Fig. 3 (a) Schematic representation of the initial chemical condition  $S_i$ . (b) Oxidation condition with respect to precursor compositions. Note that arrows represent the increment of the oxygen potential.

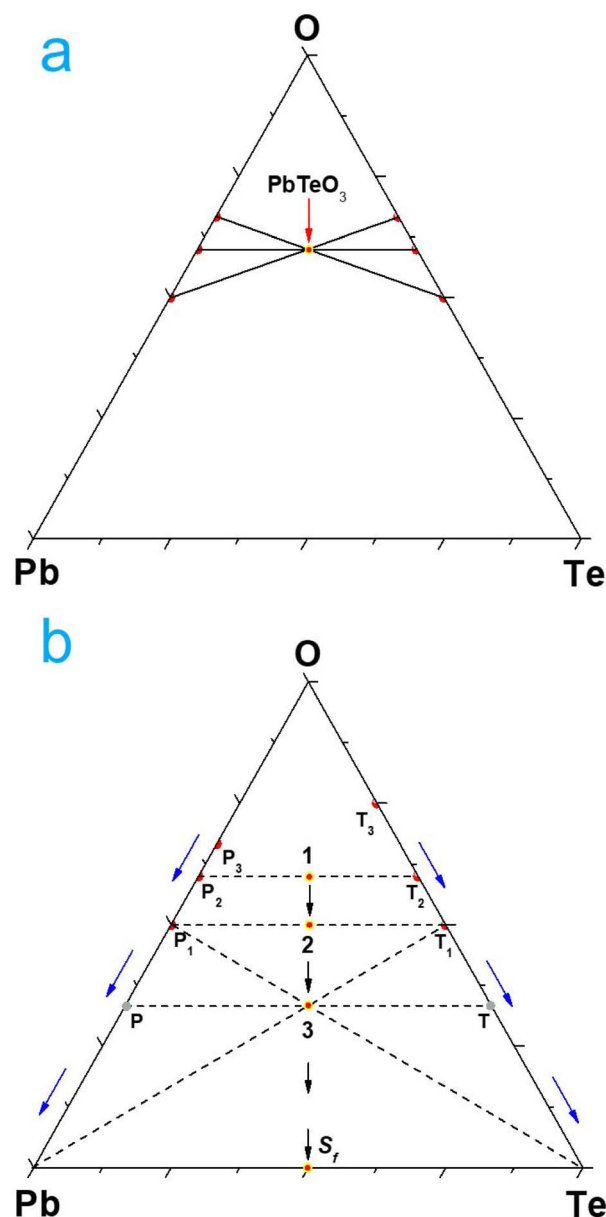


Fig. 4 (a) Nodal convergence of equivalent oxygen potential conditions represented by  $\text{PbTeO}_3$ . (b) Hypothetical reduction reaction path from 1 to  $S_f$  to transform  $\text{PbTeO}_3$  into  $\text{PbTe}$ . Note that  $S_f$  is the ending point, and arrows represent the decrement of the oxygen potential. Furthermore, note that the points P,  $P_1$ ,  $P_2$  and  $P_3$  correspond to the  $\text{Pb}_2\text{O}$ ,  $\text{PbO}$ ,  $\text{Pb}_3\text{O}_4$ ,  $\text{Pb}_2\text{O}_3$  and  $\text{PbO}_2$  oxides, while the points T,  $T_1$ ,  $T_2$  and  $T_3$  are associated with the  $\text{Te}_2\text{O}$ ,  $\text{TeO}$ ,  $\text{TeO}_2$  and  $\text{TeO}_3$  ones.





the proper interpretation of results obtained can be described as follows: in the light of experimental findings, it can be inferred that  $\text{PbTeO}_3$  is the result of the 'addition' of  $\text{PbO}$  plus  $\text{TeO}_2$ . In this context,  $\text{TeO}_2$  can only be obtained if  $\text{PbO}$  is reduced by  $\text{Te}$ . Additionally, to understand formation of  $\text{TeO}_2$  the formation, gaseous species must be considered in accordance with the thermodynamic data available in the literature.<sup>19,23,24</sup> As shown in Fig. 6,  $\text{TeO}$  (g) or  $\text{Te}_2\text{O}_2$  (g) must be regarded as an intermediate state between  $\text{TeO}$  (g)/ $\text{Te}_2\text{O}_2$  (g) to  $\text{TeO}_2$ . It is worth emphasizing that (i) these transitions are situated on the Te–O binary, (ii) the most important reactions are those where point 1 is situated and (iii) reaction among  $\text{TeO}_2$ – $\text{TeO}$  ( $\text{Te}_2\text{O}_2$ )– $\text{Te}$  should be considered in conformity with the thermodynamic data available in the literature.<sup>19,23,24</sup> Through inherent calculations, a triple point provided by the phases  $\text{PbO}$ ,  $\text{PbTeO}_3$  and  $\text{PbTe}$  will be in equilibrium at a particular temperature.<sup>19,20</sup> It is worth mentioning that equilibria change as a function of processing temperature is due to the following gaseous species: either  $\text{TeO}_2$  (g) or  $\text{TeO}$  (g)/ $\text{Te}_2\text{O}_2$  (g) versus (Te), in conformity with the thermodynamic data available in the literature.<sup>19,23,24</sup>

A straightforward way to discern this is that theoretical insights and experimental results allow us to propose that the  $\text{TeO}_2$  (gas or solid),  $\text{TeO}$  (g)/ $\text{Te}_2\text{O}_2$  (g), and  $\text{Te}_2$  (g) are a series of reduction steps encompassed in the progressive reduction of lead oxides.<sup>19,23,24</sup> To the extent that higher oxygen potentials predominate in the vial in comparison to those to trace  $\text{TeO}$  (g)/ $\text{Te}_2\text{O}_2$  (g),  $\text{PbTeO}_3$  will be associated with  $\text{PbO}$ , while lower oxygen potentials will undergo a transition where in place of interaction between  $\text{PbO}$ – $\text{TeO}_2$  a sequence will take place as follows:  $\text{Pb}$ – $\text{TeO}$  (g)/ $\text{Te}_2\text{O}_2$  (g) and  $\text{Pb}$ – $\text{TeO}$ –( $\text{Te}_2$ ). As shown in Fig. 6, the partial pressure of  $\text{Te}_2$  (g) will be in charge of reducing any Te- and/or Pb-oxides that are still present after the oxygen

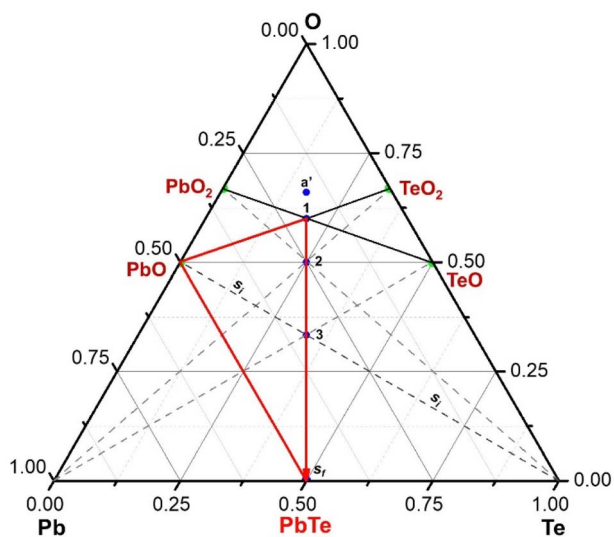


Fig. 6 Schematic chemical model depiction of the Pb–Te–O system for the mechanochemical synthesis of PbTe, considering both solid and gaseous phases. Note that (i) point 1 represents  $\text{PbTeO}_3$  stoichiometrically balanced, (ii)  $S_i$  is the starting point, and (iii)  $S_f$  is the ending point.

potential has been reduced to the smallest possible level, which will result in the formation of  $\text{PbTe}$ .<sup>19,23,24</sup>

The kinetics of converting raw materials into  $\text{PbTe}$  show that the formation of  $\text{PbTe}$  is highly favored even at the early stages of the non-equilibrium process, which is of practical significance considering that the thermodynamic and chemical mechanisms previously discussed are self-consistent with theoretical data and experimental findings.<sup>19,24</sup> However, based on thermodynamics, the presence of oxygen-containing phases predominates up until the point where oxides give way to the complete transformation of  $\text{PbTe}$ , taking into consideration the occurrence of the triple phase unalterable point,  $\text{PbO}$ – $\text{PbTeO}_3$ – $\text{PbTe}$ . It is essentially important to note how the Pb–Te–O system's phases are grouped; this triangular representation illustrates a high degree of symmetry between oxides and  $\text{PbTe}$ . Consequently, information on this model illustrates how two types of heterogeneous oxides are associated with the  $\text{PbTe}$ .

A comment worth mentioning is that physical-metallurgical stimuli, such as plastic deformation and structural defects, should be considered as stored energy during the non-equilibrium process. Chemical crystal transitions, which may be primarily considered as entropic changes in the system, could be encouraged by such stimuli. In terms of mechanochemical stimuli, deformed  $\text{PbO}$  in close contact with  $\text{Te}$  causes a reduction of the former phase in restricted locations, including the nano- and microscale, due to the distinct mechanical reactions of the reactant particles. A transition balance between  $\text{PbO}$ ,  $\text{Te}$ ,  $\text{TeO}$  (or  $\text{Te}_2\text{O}_2$ ), and  $\text{TeO}_2$  (gas or solid) would be established attributable to the negative free energy of formation of the Te-enriched phases,<sup>19,20,24</sup> revealing  $\text{PbTeO}_3$  as an intermediate as-milled reaction product. Nevertheless, the gaseous tellurium overrides the partial pressure of oxygen during the milling process, giving rise to  $\text{PbTe}$ .<sup>19,20,23,40</sup>

The O– $\text{PbTe}$  path allows for a schematic representation of how, in theory, it may progress from the 1 point to the  $S_f$  one once the highest oxygen chemical potential is achieved. Thus, the stages through these transitions are defined as follows:  $\text{PbTeO}_3 \rightarrow \text{PbTeO}_2 \rightarrow \text{PbTeO} \rightarrow \text{PbTe}$ , as proposed in the literature.<sup>20,29</sup>

### The PbO–Te pseudo-binary system

To supplement the schematic chemical model representation of the Pb–Te–O system for obtaining  $\text{PbTe}$ , a graphic depiction of the pseudo-binary  $\text{PbO}$ – $\text{TeO}_2$  local equilibrium is proposed to enrich the understanding of the mechanochemical synthesis of  $\text{PbTe}$ , see Fig. 7. In this graphic representation  $A_1$ ,  $C_1$  and  $F_1$  have been unambiguously detected by the XRD technique.<sup>19,20,24</sup> Additionally, with the same technique supplemented by SIMS,<sup>41</sup>  $C_5$  has been extensively identified, and this result has also been confirmed by the EDS elemental map technique.<sup>28</sup> In what surface characterization is concerned,  $C_1$ ,  $C_4$ ,  $C_5$  and  $F_1$  have been detected by EDX-STEM, SAED and XPS techniques.<sup>19,28,41</sup> It is worth highlighting that  $B_1$ ,  $D_1$ , and  $E_1$  have not been detected by any experimental technique during milling, but  $B_1$  and  $E_1$  have been reported in the literature.<sup>42,43</sup>



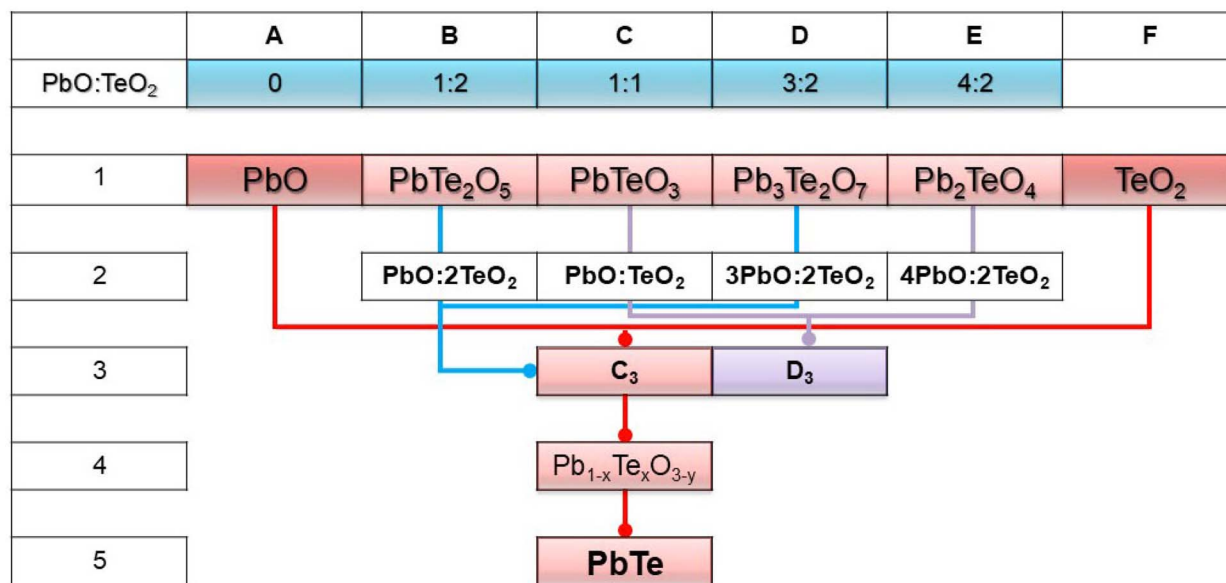


Fig. 7 Schematic depiction illustrating the chemical interactions in the pseudo-binary PbO–TeO<sub>2</sub>. Note that the as-milled products are placed as stoichiometric, non-stoichiometric, experimental, and theoretical or even undetected complex oxides. The PbO : TeO<sub>2</sub> row indicates the molar ratio of the intermediate reaction products.

As shown in Fig. 7, it illustrates that the pseudo-binary PbO–TeO<sub>2</sub> local equilibrium makes it evident how the addition of its contiguous phases yields intermediate oxides. As a result, it is feasible to recognize a set of pair interactions between A and F, which enable the formation of intermediate oxides, specifically C<sub>3</sub> and D<sub>3</sub>. Note that the 1 point—in Fig. 6—is represented as C<sub>3</sub> in Fig. 7. It is worth mentioning that C<sub>3</sub> can be obtained in two diverse ways. In the first case, by the addition of its adjacent phases (B<sub>1</sub> and D<sub>1</sub>) and second, by adding the extremes (A<sub>1</sub> and F<sub>1</sub>). Another intriguing feature is that D<sub>1</sub> is not governed by the addition of its adjacent phases, as previously established.

Fig. 7 schematically illustrates the decrease in oxygen chemical potential in the oxides between A<sub>1</sub> and F<sub>1</sub> (row one → row three → row four → row five). In this regard, a more thorough analysis revealed that rows 1 and 2 correspond to equivalent phases in Fig. 7. Row 3 suggests that when the PbO : TeO<sub>2</sub> ratio is equal to 1.0, PbTeO<sub>3</sub> can be formed. Furthermore, the same result can be achieved when the undetected phases (B<sub>1</sub> and D<sub>1</sub>), which represent the maximum oxygen chemical potential, are added. It is key to point out that, in this latter case, the PbO : TeO<sub>2</sub> ratio is also equal to 1.0.

Finally, it is expected that the oxygen potential decreases as follows: PbTeO<sub>3</sub> → Pb<sub>1-x</sub>Te<sub>x</sub>O<sub>3-y</sub> → PbTe (C<sub>1</sub> → C<sub>4</sub> → C<sub>5</sub>), noting that this trend—in Fig. 7—is located along the O–PbTe direction in Fig. 6. It is worth noting that the high-energy collisions, during milling, give way to c–P–T vial conditions, which change the rate and outcome of the phase transformations, as the milling time proceeds, and may result in the appearance of unusual states or phases, distinct to those expected under typical equilibrium processing conditions.<sup>29</sup> The Pb<sub>1-x</sub>Te<sub>x</sub>O<sub>3-y</sub> is an example of those unusual phases; it is an amorphous or non-stoichiometric phase which is formed under the following condition:  $pO_2(\text{PbTeO}_3) > pTe_2$ .<sup>19,23</sup>

In simple terms, due to chalcogen possessing close chemical similarities, because the elements in the same group of the periodic table have similar chemical properties, they have been treated as “rock-salt IV–VI semiconductors” since their characteristics do not differ from each other. Therefore, as shown in the ESI,<sup>†</sup> considering equivalent systems (*e.g.*, Pb–Se–O) as well as processing parameters, reasonable analogous chemical models can be generated, including the formation of ternary semiconductor solid solutions and II–VI semiconductors (*e.g.*, ZnTe).

## Conclusions and outlook

A chemical model has been constructed to map the transformation from precursors to PbTe semiconductor material at non-equilibrium processing conditions under an air atmosphere, taking milling time into consideration. The interpretation of the chemical model allows us to visualize the chemical pathway from precursors to the semiconductor without the need for specialized software, facilitating understanding, in detail, about the obtaining of PbTe semiconductors through non-equilibrium process conditions.

The chemical model permits a closer look at the chemical interactions among the phases, facilitating phase correlation during milling. In fact, to supplement the schematic chemical model representation of the Pb–Te–O system for the mechanochemical synthesis of PbTe, a graphic representation of the pseudo-binary PbO–TeO<sub>2</sub> local equilibrium was proposed to provide a more complete picture.

An important outcome is that the chemical model, based on the Gibbs composition triangle, provides enough experimental and theoretical data to understand the synthesis path related to the non-equilibrium process. Furthermore, the usage of such



chemical models is not limited to the proposed system. Additionally, considering equivalent systems (e.g., Pb–Se–O) as well as processing parameters, reasonable analogous chemical models can be generated, including the formation of ternary semiconductor solid solutions. Therefore, this proposal presents new opportunities for visualizing the chemical pathway during the synthesis of semiconductors under non-equilibrium conditions.

## Data availability

Data will be made available on request.

## Author contributions

Conceptualization, investigation, formal analysis, visualization, writing – original draft, writing review and editing: H. Rojas Chávez.

## Conflicts of interest

There are no conflicts to declare.

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