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## Effect of silver and cobalt on transparent conducting CdO thin films: tuning the optoelectronic properties†

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Thin film technology is a crucial area of research for the advancement of modern technologies and next-generation applications. In this work, the structural, optoelectrical, and thermal properties of silver (Ag)-cobalt (Co) dual-doped cadmium oxide (CdO) thin films were investigated. Successful incorporation of the doping elements with high crystallinity (maximum value 85%) was confirmed through X-ray diffraction (XRD) analysis, and no unexpected peaks were observed. The pure and doped samples demonstrated a polycrystalline nature with FCC lattice. Scanning electron microscopy (SEM) results revealed the surface microstructures of the deposited 3% Ag-Co doped CdO thin films as spherical or round-shaped agglomerated grains with a smooth surface appearance. From the optical analysis, a maximum transmittance of 89% was observed for the 3% Ag-Co doped CdO sample, which was rarely reported in the literature. Additionally, doping altered the optical band gap from 4.09 eV to 3.88 eV, which enhanced the potential of using the transparent conductive films in optoelectronic devices. Doping CdO with Ag and Co altered its electrical properties, leading to an increase in resistivity and a decrease in carrier concentration. Hall measurements confirmed that all the samples showed an n-type semiconductor behavior. Thermal analysis results demonstrated that the 1% Ag-Co doped CdO sample exhibited the highest thermal conductivity of  $2.35 \text{ W m}^{-1} \text{ K}^{-1}$ , and further increase in the doping concentrations led to a decrease in both the power factor and figure of merit ( $ZT$ ) values.

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## 1 Introduction

Thin film technology (TF) encompasses a diverse range of materials and methods that have contributed to the advancements of applications across various fields, including electronics, photovoltaics, gas sensing, and energy conversion and storage. Advanced TF technology has become an important area of research in the development of transparent conductive oxides (TCOs), which are essential components in various optoelectronic devices as they have impressive and unique

characteristics of high optical transparency and good electrical conductivity.<sup>1,2</sup> Among various TCOs, aluminium-doped zinc oxide (AZO),<sup>3</sup> indium-doped tin oxide (ITO),<sup>4</sup> and fluorine-doped tin oxide (FTO)<sup>5</sup> have both good electrical conductivity and optical transparency, making them useful in solar cells,<sup>2</sup> flat panel displays,<sup>6</sup> and light-emitting diodes (LEDs).<sup>7</sup> The ability to control the chemical composition during deposition of the films makes TCOs of binary compounds more promising than the multicomponent oxides,<sup>1</sup> such as  $\text{Cd}_2\text{SnO}_4$ ,  $\text{CdSnO}_3$ , and  $\text{CdIn}_2\text{O}_4$ .<sup>8,9</sup>

CdO is one of the most promising binary TCO materials with widespread applications in optoelectronic and electronic applications owing to its excellent electrical conductivity, good optical transparency throughout the visible spectrum, and impressive carrier density and mobility.<sup>10</sup> These remarkable characteristics make CdO a functional choice for diverse applications, such as solar cells, phototransistors, diodes, gas sensors, and antireflection coatings.<sup>11</sup> Besides, its optical band gap, ranging from approximately 2.2 eV to 2.5 eV, enables high transmittance in the visible spectrum, making CdO a strong TCO suitable for diverse electronic applications, such as solar cells, and photodiodes.<sup>11-13</sup> Lavate *et al.*<sup>14</sup> demonstrated the

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photodegradation capabilities of CdO thin films, indicating its potential for applications in environmental remediation. This characteristic suggests the applicability of CdO thin films beyond its traditional electronic applications.

Doped CdO thin films have become a pivotal area of research due to their promising properties for various applications, particularly in optoelectronics and gas sensing applications. For instance, doping diverse metallic ions into the CdO lattice can enhance its properties.<sup>15</sup> Metal doping can alter the optical properties of materials. The influence of the metal's interaction with electromagnetic waves is based on the free conduction of electrons in the metal. High reflectance may be observed at optical frequencies due to the presence of a negative dielectric constant in many metals.<sup>16</sup> Surface plasmon resonance (SPR) is a phenomenon that occurs when P-polarized light hits a metal-coated prism, creating free electrons in the metal surface that can take part in certain interactions, resulting in a collective oscillation known as surface plasmon resonance (SPR).<sup>17</sup> The noble metal gold, known for its high oxidation resistance, is mainly used to excite SPR;<sup>18</sup> however, given its high cost, other noble materials, such as silver, may also be used to generate SPR.<sup>19</sup> Generally, a lower thickness of film generates the maximum SPR effect.<sup>20</sup>

The electrical properties of thin films cover a wide area due to the different microstructures resulting from different doping materials and concentrations, whether they are metallic films, semiconductors, or insulator films, and the type of substrate on which they are deposited. Doping metallic ions in CdO whose ionic radii are less than that of Cd<sup>2+</sup> can improve the electrical conductivity.<sup>21</sup> Even with ions with a small radius, the many structural defects and grain boundaries in microstructures can act as obstacles to charge carriers, thereby reducing the conductivity.<sup>22</sup> These defects can trap mobile carriers and generate an energy barrier between the crystallites by decreasing the motion of the charge carriers.<sup>23</sup>

Tin (Sn)-doping of CdO thin films exhibited enhanced conductivity that was 2–5 times higher than that of ITO coatings.<sup>24</sup> It was also reported that Y- and In-doping into CdO could alter the transparency and electronic structure of CdO thin films, making them suitable for optoelectronic device applications.<sup>25</sup> Neodymium (Nd)-doped CdO exhibited a lower optical band-gap with increasing the Nd concentration, which is important for solar cells and photodetector applications.<sup>26</sup> Therefore, the incorporation of suitable dopants into the CdO lattice can significantly influence the structural, optical, and electrical characteristics of CdO films, making them suitable for an extensive range of technological applications. Many studies have reported on doping CdO, including with aluminum (Al),<sup>27</sup> silver (Ag),<sup>28</sup> copper (Cu),<sup>29</sup> zinc (Zn),<sup>30</sup> molybdenum (Mo),<sup>31</sup> and iron (Fe).<sup>32</sup> Many studies have also sought to control the optical and electrical properties of CdO through dual doping with different elements, including Ni-Ga,<sup>33</sup> Cu-Fe,<sup>34</sup> Zn-Co,<sup>35</sup> and Cr-Co.<sup>36</sup>

Previous studies have emphasized that dual doping can lead to a significant enhancement of the properties of CdO, improving its potential for use in various optoelectronic applications.

Mn-Fe dual-doped CdO exhibited improved transmittance due to the Moss–Burstein effect.<sup>37</sup> Co-doped CdO thin films with Zn and Ag demonstrated outstanding non-linear optical properties, improved conductivity and, carrier concentrations.<sup>38</sup> Tb-Zn dual-doped CdO thin films showed greatly elevated electrical properties with increasing the Tb concentration.<sup>39</sup> The incorporation of Zn and Co in CdO thin films promoted the optical transmittance and ferromagnetic properties, which are suitable characteristics for spintronic applications.<sup>35</sup> Also, the magnetic properties of CdO thin films were improved by dual doping with Cu and Fe.<sup>34</sup> CdO thin films could also attain ferromagnetic properties with Ni-Ga dual doping,<sup>33</sup> while Ni-Zn dual doping led to a ferromagnetic transition from paramagnetic, holding promise for a wide range of spintronic applications.<sup>40</sup> Among all the potential doping elements, Ag is perhaps the most impressive doping element because of its ability to remove the metal-like behavior of CdO, while increasing the Ag concentration in CdO can enhance both the electrical and optical properties, including reducing the band-gap energy and refractive indices, as reported by Alam *et al.*<sup>41</sup> Zhang *et al.*<sup>42</sup> found that Ag-doping enhanced the near-band edge UV emission in metal oxide films. It was also found that metal doping could increase the conductivity in TCO,<sup>43</sup> but excessive metal dopant levels could have a negative impact on the conductivity in TCO because of the introduction of possible defects<sup>44</sup> and its lower crystallinity.<sup>21</sup> Additionally, the electrical conductivity could also be increased with increasing Ag concentration.<sup>45</sup> On the other hand, co-doping helps enhance the optical and electrical properties. It was reported that increasing the cobalt content led to the optical band gap decreasing,<sup>46</sup> and the optical transmittance increasing.<sup>47</sup> Besides, Co ions can introduce ferromagnetism into the CdO matrix, which would make it suitable for spintronic applications where both the magnetic and electronic functionalities of materials are required.<sup>48</sup>

However, the consequences of the dual doping of Ag and Co with CdO remain largely unexplored. Motivated by all the prior findings in this field, this research aimed to investigate the dual-doping of Ag and Co with CdO thin films. Demet *et al.*<sup>49</sup> found that the optical transmittance of CdO thin films increased with increasing the Co concentration from 1% to 3%. The optical band-gap and refractive index were found to decrease with increasing the Ag concentration from 1% to 4%.<sup>41</sup> Khan *et al.*<sup>50</sup> reported a significant improvement in the optical properties at a higher Ag concentration (4%). Motivated by these findings, we aimed to dope both silver (Ag) and cobalt (Co) in equal percentages ranging from 1% to 4%. Here, all the films were deposited on a glass substrate using spray pyrolysis at 400 °C. We utilized the spray-pyrolysis deposition method for film fabrication, while many previous studies synthesized pure and doped CdO utilizing other methods, including pulse laser deposition,<sup>51</sup> spray pyrolysis,<sup>52</sup> successive ionic layer adsorption and reaction (SILAR),<sup>53</sup> magnetron sputtering,<sup>54</sup> sol-gel spin coating,<sup>55</sup> and chemical vapor deposition.<sup>56</sup> Among these methods, the spray-pyrolysis method is cost-effective and has the unique feature of allowing precise control over the film



thickness and doping concentrations. After fabrication of the Ag–Co dual-doped CdO thin films by applying the spray-pyrolysis method, we analyzed the structural, morphological, optical, electrical, and thermal properties of the Ag–Co dual-doped CdO thin films with a view to their application potential. The findings of this research should greatly contribute to the advancement of the general efficiency of optoelectronic devices.

## 2 Experimental section

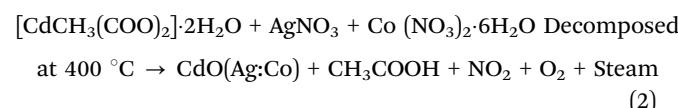
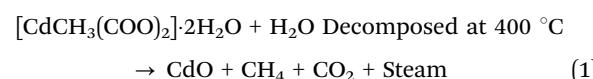
### 2.1 Materials

The materials used for the deposition of the thin films included cadmium acetate dihydrate ( $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ ) as a source for CdO, silver nitrate ( $\text{AgNO}_3$ ) for Ag, and cobalt(II) nitrate hexahydrate ( $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ) for Co. These chemicals were sourced from Merck KGaA with purity levels of 99%, 99.8%, and 98%, respectively. The solvent was deionized water, and the substrates used were micro-glass slides.

### 2.2 Deposition of the thin films

Deposition of the CdO thin films on glass substrates was performed using the spray-pyrolysis (SP) method under normal atmospheric conditions, as illustrated in Fig. 1. Both undoped and co-doped CdO thin films were prepared by applying a solution created from separately prepared 0.1 M solutions of each substance dissolved in deionized water. The solutions were mixed using a magnetic stirrer. The compositions of the sprayed solutions are provided in Table 1. A heating plate served as the heat source with a temperature of 400 °C. The chemical solutions were applied on the glass substrates using a nozzle as a spraying apparatus. Prior to the deposition of the thin films, the glass substrates were cleaned with deionized water and organic solvent and then left to dry under normal atmosphere conditions. Next, these glass substrates were slowly heated before placing them on the heating plate, which contributed significantly to minimizing the thermal shock. The distance

of the substrate from the atomizer was kept constant at 26 cm. An air compressor operating at 1 bar pressure was employed for each sample, and a liquid flow rate of  $1 \text{ ml min}^{-1}$  was utilized. The spray nozzle ensured a uniform deposition of the solution over the substrate, and the time taken for the spraying was 20 min, with a 5-min interval between sprays. The most probable reactions during the thin film deposition process are as given in the following eqn (1) and (2).



### 2.3 Characterization and film analysis

In this study, an X-ray diffractometer (Empyrean EMP 3, Malvern Panalytical) was used to investigate the structural characteristics of the CdO and Ag–Co dual-doped thin films.  $\text{CuK}\alpha$  ( $K = 1.5406 \text{ \AA}$ ) X-ray radiation was used to scan every sample between diffraction angles of  $20^\circ$  and  $80^\circ$  at room temperature. The Debye–Scherrer equation (eqn (3)) was utilized to calculate the average sizes of the CdO and CAC crystallites.<sup>57</sup>

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (3)$$

where  $D$  is the crystallite size,  $\lambda$  is the X-ray wavelength ( $1.5406 \text{ \AA}$ ),  $k$  is Scherrer's constant and is equal to 0.9 for spherical crystals (wurtzite/cubic),  $\beta$  is the full width at half maximum (FWHM), and  $\theta$  is the Bragg's angle in radians. Here,  $k$  depends on the crystallite shape and the size distribution, indices of the diffraction line, and the actual definition used for  $b$ , whether FWHM or an integral breadth.<sup>58</sup> The peak intensities related to the (111) and (200) planes were used to determine the lattice constant parameter ( $a$ ), cell volume ( $V$ ),

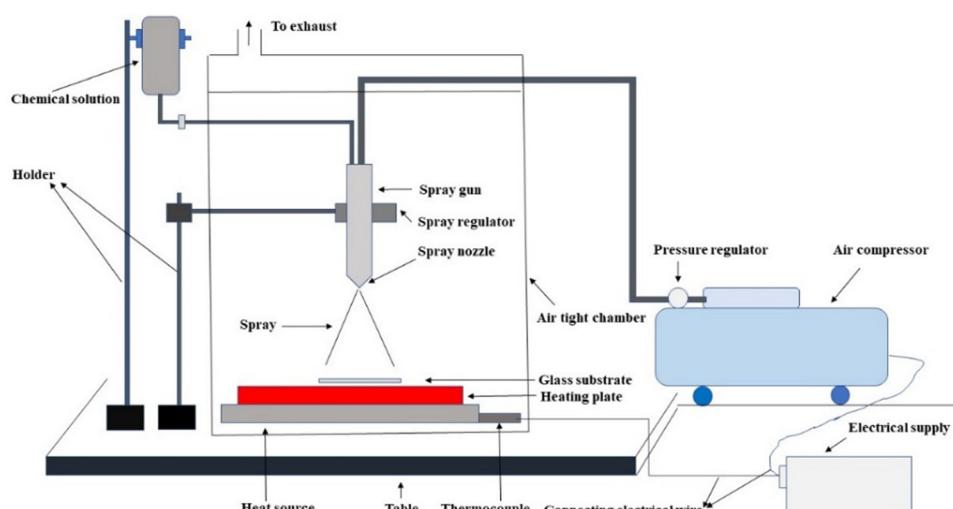


Fig. 1 Schematic of the experimental setup.



Table 1 Compositions of the sprayed solutions

Sample	Solution composition
CdO	100% $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ solution
CAC-1	1% $(\text{AgNO}_3)$ solution + 1% $(\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O})$ solution + 98% $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ solution
CAC-2	2% $(\text{AgNO}_3)$ solution + 2% $(\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O})$ solution + 96% $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ solution
CAC-3	3% $(\text{AgNO}_3)$ solution + 3% $(\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O})$ solution + 94% $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ solution
CAC-4	4% $(\text{AgNO}_3)$ solution + 4% $(\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O})$ solution + 92% $[\text{CdCH}_3(\text{COO})_2] \cdot 2\text{H}_2\text{O}$ solution

strain ( $\varepsilon$ ), and dislocation density ( $\delta$ ), using the following relations (eqn (4)–(7)).<sup>59,60</sup>

$$d = \frac{a}{\sqrt{h^2 + k^2 + l^2}} \quad (4)$$

$$V = a^3 \quad (5)$$

$$\varepsilon = \frac{\beta \cos \theta}{4} \quad (6)$$

$$\delta = \frac{1}{D^2} \quad (7)$$

The crystallinity is the proportion of crystalline areas within a material and represents the extent of long-range order in a crystalline substance.<sup>61</sup> For measurement of the crystallinity, the total crystalline peak area and the total area (crystalline and amorphous area) were measured using Origin software, and the following equation (eqn (8)) was used to determine the crystallinity.

$$\text{Crystallinity (\%)} = \left( \frac{\text{Crystalline area}}{\text{Crystalline area} + \text{Amorphous area}} \right) \times 100 \quad (8)$$

The surface morphology of the CdO and Ag–Co dual-doped CdO thin films was characterized by SEM (TESCAN VEGA Compact). Additionally, elemental composition analysis was conducted by energy-dispersive X-ray spectroscopy (EDS) with the same instrument (TESCAN VEGA Compact).

The thickness of the thin film samples was measured using a BRUKER surface profilometer (DktakXT).

A UV-visible spectrophotometer SHIMADZU (Uv-1900i) was used to analyze the optical transmittance (T) and absorbance (A) of the CdO and (Ag–Co) dual-doped CdO thin films within the wavelength range of 350–1100 nm at 25 °C. The optical band gap of the samples was determined using eqn (9), which is commonly known as Tauc's law, where  $\alpha$  is a coefficient,  $A$  is a constant,  $h$  is Plank's constant,  $\nu$  is the absorption at a specific frequency of light,  $E_g$  is the optical band gap, and  $n$  is a constant.<sup>62</sup>

$$\alpha h\nu = A(h\nu - E_g)^n \quad (9)$$

The absorption coefficient values, extinction coefficient ( $k$ ), and refractive index ( $n_r$ ) were determined using eqn (10)–(12), respectively. In these equations,  $T$  represents the transmittance,  $d$  is the film thickness,<sup>63</sup>  $\lambda$  represents the wavelength

of light, and  $r$  denotes the reflectance.

$$a = \frac{\ln \frac{1}{T}}{d} \quad (10)$$

$$k = \frac{a\lambda}{4\pi} \quad (11)$$

$$n_r = \frac{1+r}{1-r} \sqrt{\frac{4r}{(1-r)^2} - k^2} \quad (12)$$

The Van der Pauw Ecopia HMS-3000 Hall Measurement System was used for electrical analysis. The thin film samples' conductivity and resistivity were determined using eqn (13), where  $I$  is the current, and  $R$  is the resistance.<sup>64</sup>

$$V = IR \quad (13)$$

The Hall effect was measured at room temperature using a constant magnetic field of 0.545 T. The determined Hall voltage ( $V_H$ ) was utilized to compute the Hall coefficient ( $R_H$ ), carrier concentration ( $n$ ), and Hall mobility ( $\mu_H$ ) using eqn (14)–(16).<sup>25,65</sup>

$$R_H = \frac{V_H t}{IB} \quad (14)$$

$$\mu_H = \frac{R_H}{\rho} \quad (15)$$

$$n = \frac{1}{R_H e} \quad (16)$$

A thermal conductivity meter (HOT DISK TPS 500S) was used to determine the thermal conductivity, diffusivity, and specific heat of the undoped and Ag–Co dual-doped CdO samples. The power factor (PF) was calculated from eqn (17), where  $S$  is the Seebeck coefficient and  $\rho$  is the electrical resistivity.

$$\text{PF} = \frac{s^2}{\rho} \quad (17)$$

The performance of a thermoelectric material is called the figure of merit, which can be determined using the following equation (eqn (18)), where  $S$  is the Seebeck coefficient,  $T$  is the absolute temperature,  $\rho$  is the electrical resistivity, and  $K$  is the thermal conductivity.

$$ZT = \frac{s^2 T}{\rho K} \quad (18)$$

### 3 Result and discussion

#### 3.1 X-ray diffraction (XRD) analysis

The diffraction peaks observed for the (111), (200), (220), (311), and (222) planes confirmed the face-centered cubic (FCC) structure of the CdO, which was consistent with the standard JCPDS card no. 05-0640. These peaks indicated a polycrystalline structure with a face-centered cubic (FCC) lattice. The absence of additional peaks suggested that there were no secondary phases related to the dopants or their oxides, confirming the successful integration of Co and Ag into the CdO lattice. Upon doping with silver, a shift in the preferred orientation from the (200) plane to the (111) plane was observed, with a similar shift reported by Alam *et al.*<sup>41</sup> This shift, along with the slight rightward peak position noted in the XRD patterns, provided further evidence of the successful Co and Ag doping, consistent with Velusamy *et al.*'s findings.<sup>66</sup> The CdO thin film's lattice parameter was found to be very close to the standard value (4.695 Å) as per JCPDS card no. 05-0640 (Fig. 2).

The crystallite sizes were calculated using the Debye–Scherer formula (eqn (3)). The crystallite sizes for the (111) plane ranged from 14.65–43.98 nm (Table 2). The dual-doped samples exhibited larger crystallite sizes compared to the undoped CdO films (Fig. 3). This increase could be attributed to the larger ionic radii and atomic weights of the dopants' compared to cadmium ions, a phenomenon also observed by Seçkin *et al.*<sup>67</sup>

However, the CAC-3 sample exhibited a decrease in crystallite size, likely due to the increased lattice strain and defects caused by incorporating the different-sized dopants, which prevented the crystallites from growing as large as in the CAC-4 sample. The slight increase in crystallite size with Ag-doping was attributed to Ag's substitution of oxygen vacancies at the grain boundaries, which contributed to a reduction in the local strain.<sup>41</sup>

The dislocation density ( $\delta$ ), a measure of defects in the crystal structure (eqn (7)), and the lattice strain ( $\varepsilon$ ) (eqn (6)) results indicate that the crystallinity decreased with increasing the doping concentration. This decrease could be attributed to the size differences between Cd and Ag ions, which disrupted the crystalline order, a trend consistent with the observations made by Khan *et al.*<sup>68</sup> The highest crystallinity was observed in the undoped CdO film, while the lowest was in the 3% doped sample CAC-3, likely due to the introduction of significant lattice strain and defects. Upon doping, a reduction in peak intensity was observed, particularly for the (200) and (111) planes, further confirming the successful incorporation of Ag and Co into the CdO lattice, as these dopants introduced strain and disorder into the crystal structure, reducing the overall crystallinity.

#### 3.2 Scanning electron microscopy (SEM) and energy-dispersive spectroscopy (EDS) analyses

SEM is an effective method for analyzing the surface morphology of thin films, and it is widely acknowledged that the surface structure plays a crucial role in device technology applications.<sup>69–73</sup> The surface microstructures of the deposited CdO thin films obtained from SEM analysis were compared in Fig. 4. According to the SEM microstructures, all the films had spherical or rounded-shaped agglomerated grains. The pure CdO, CAC-1, and CAC-4 thin films comprised rounded-shaped grains and a few patches where the particles had agglomerated more densely. The CAC-2 and CAC-3 thin films displayed uniformly dispersed, spherical-shaped grains and some tiny agglomerations that were homogeneous. According to the precise analysis with ImageJ software, the CAC-2 thin film contained tiny voids inside its particles, which decreased with increasing the Ag–Co concentration. The average particle size also decreased with increasing the doping concentration, with

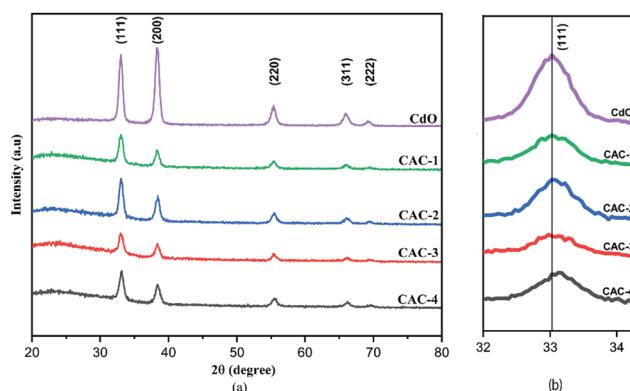


Fig. 2 (a) X-ray diffraction (XRD) patterns and (b) peak shifting (111) plane of the pure CdO and Ag–Co dual-doped CdO thin films.

Table 2 Obtained structural XRD parameters of the pure CdO and the Ag–Co dual-doped CdO thin films along the (111) and (222) planes

Composition	( <i>hkl</i> )	2θ	FWHM	Crystallite size, <i>D</i> (nm)	<i>d</i> -spacing, (Å)	Lattice parameter, <i>a</i> , (Å)	Unit cell volume	Dislocation density $\delta \times 10^{-3}$ (nm <sup>-2</sup> )	Lattice strain ( $\varepsilon \times 10^{-3}$ )	Crystallinity (%)
CdO	(111)	33.023	0.590	14.658	2.710	4.694	103.432	4.654	10.277	85.37
CAC-1		32.927	0.393	21.981	2.717	4.700	103.850	2.069	6.851	64.36
CAC-2		33.033	0.196	43.976	2.709	4.687	102.990	0.517	3.425	69.73
CAC-3		33.029	0.590	14.658	2.709	4.687	103.023	4.653	10.277	56.60
CAC-4		33.117	0.246	35.188	2.702	4.681	102.58	0.807	4.282	65.02
CdO	(200)	38.274	0.639	13.731	2.349	4.699	103.769	5.303	11.124	85.37
CAC-1		38.315	0.196	44.815	2.347	4.695	103.531	0.497	3.408	64.36
CAC-2		38.502	0.196	44.658	2.336	4.672	102.017	0.501	3.422	69.73
CAC-3		38.357	0.590	14.879	2.344	4.689	103.131	4.516	10.268	56.60
CAC-4		38.391	0.393	22.321	2.342	4.685	102.871	2.006	6.845	65.02



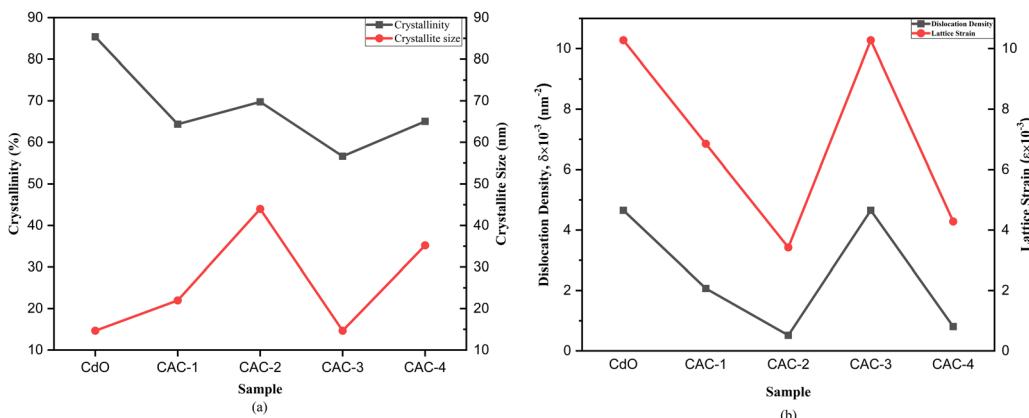


Fig. 3 Variation of the (a) crystallinity and crystallite size and (b) dislocation density and lattice strain of the pure CdO and Ag–Co dual-doped CdO thin films.

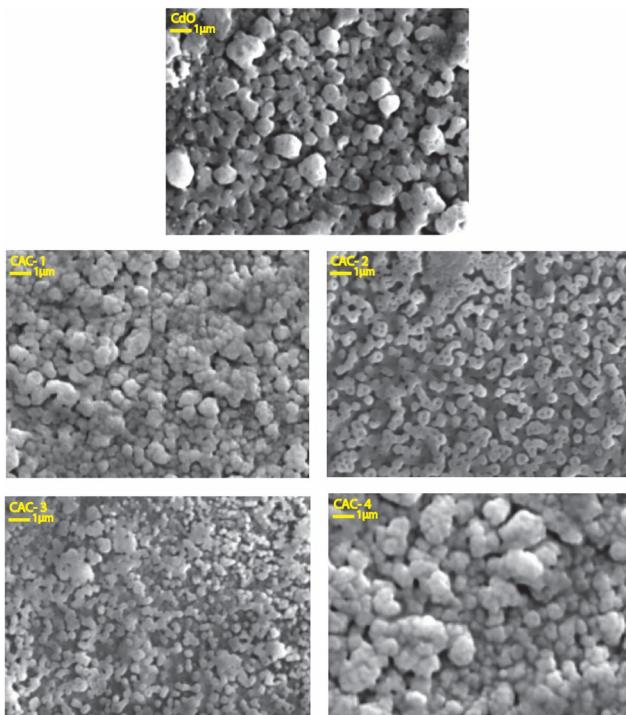


Fig. 4 Scanning electron microscopy images of the pure CdO and Ag–Co dual-doped CdO thin films.

the CAC-3 thin film showing the lowest average particle size, *i.e.*, 348 nm, leading to its smoother surface appearance compared to in the other SEM images.

Elemental composition analysis of the pure CdO, CAC-1, CAC-2, CAC-3, and CAC-4 thin films was performed using energy-dispersive spectroscopy (EDS) (Fig. 5). The EDS results confirmed the presence of cadmium, oxygen, silver, and cobalt elements in the produced Ag–Co dual-doped CdO thin films. On the other hand, Cd and O were the only two elements in the pure CdO sample. Ag and Co peaks appeared after doping the CdO thin films with 1%, 2%, 3%, and 4% Ag and Co ions. This indicates

that the Ag and Co ions were successfully incorporated into the CdO lattice. As the Ag and Co percentage increased, the atomic percentage of O in the prepared CdO films decreased from 83.94% to 59.92%. The measured atomic and weight percentage ratios of Cd, O, Ag, and Co verified the presence of Ag and Co in the spray-deposited CdO thin films (Table 3).

### 3.3 Optical analysis

This research observed that the optical behavior of the CdO thin films was influenced by the dual-doping elements Ag and Co introduced into the CdO structure, which improved the material's optical properties.

Fig. 6(a) shows the transmittance spectra of the undoped and dual-doped CdO thin films. Except for the CAC-3 sample, the transparency of the dual-doped films was lower than that of the pure CdO thin film. The distortion produced by the dual-doped ions,<sup>35</sup> the greater crystallite size,<sup>25</sup> and the lower surface roughness<sup>74</sup> may be responsible for the decreasing transmittance values. A higher transmittance value of 89% was found in the CAC-3 sample, and the maximum transparency range of 86%–89% stayed in the visible and near-IR region. XRD also showed that the crystallite size of the CAC-3 sample was the lowest (14.65 nm). So, the XRD results strongly support the highest transmittance value of this sample. A sample with a high transmittance value is suitable for functioning in the active region of solar cells.<sup>75</sup>

The extinction coefficient is related to a material's light-absorbing capacity, and this value was greater for all the dual-doped films than that of the pure CdO thin film (Fig. 6(b)). However, the overall extinction coefficients of all the samples all were less than 0.1. This type of diminished extinction coefficient is suitable for the window layer to transmit light efficiently.<sup>75</sup>

Fig. 6(c) shows the wavelength-dependent refractive index of the pure and dual-doped CdO thin films. A greater refractive index was found for the lower dual-doped films, and decreased with increasing the doping concentration. Therefore, the dual-doped concentration in the CdO thin film affected the sample's



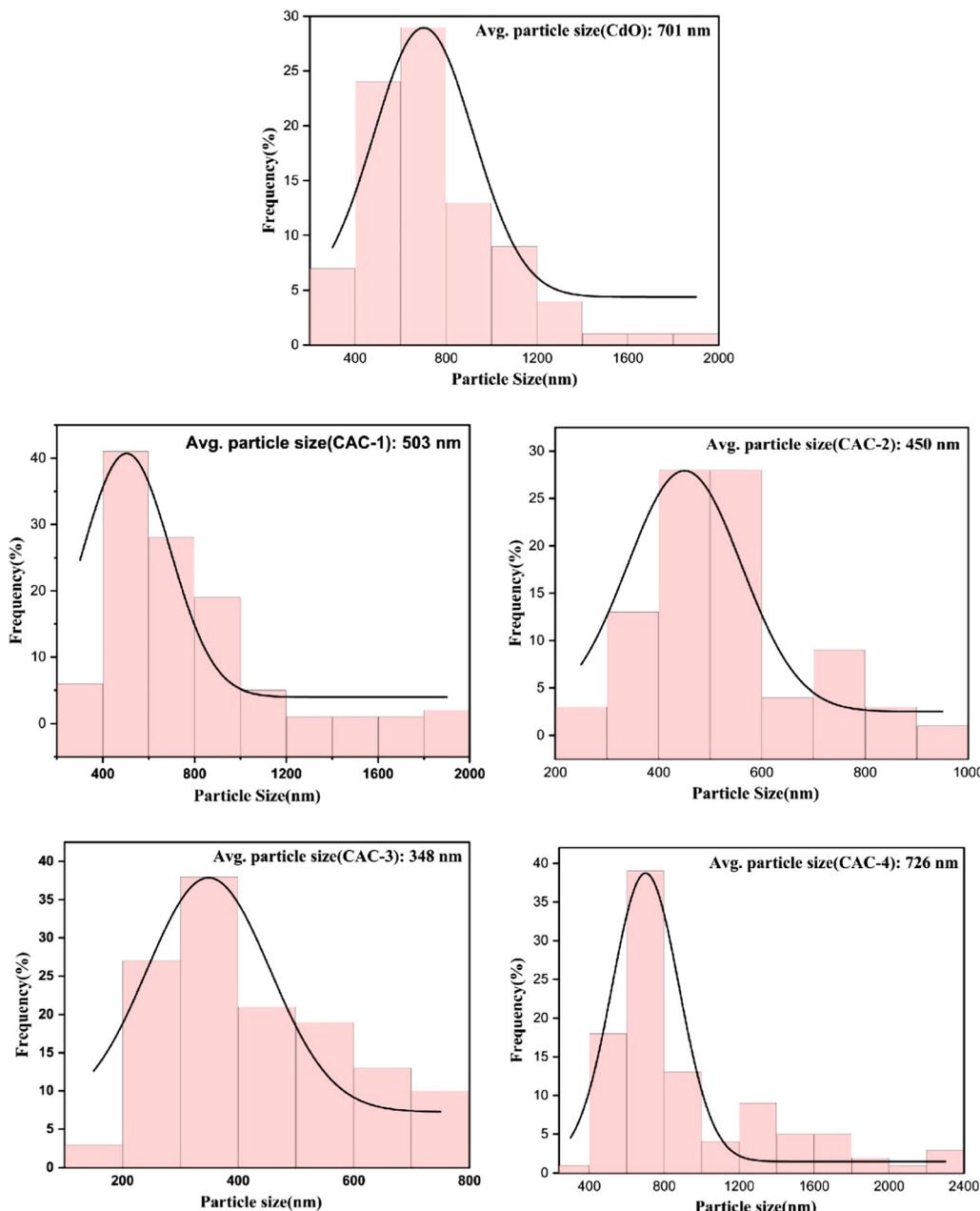


Fig. 5 Average particle sizes obtained from the histogram images of pure CdO and Ag-Co dual-doped CdO thin films.

Table 3 Elemental information obtained from EDS in terms of atomic (%) and weight (%) of the pure CdO and Ag-Co dual-doped CdO thin films

Sample	Atomic (%)			Weight (%)				
	Cd	O	Ag	Co	Cd	O	Ag	Co
CdO	16.06	83.94	—	—	57.34	42.66	—	—
CAC-1	19.43	77.89	1.6	1.08	59.56	33.99	4.72	1.73
CAC-2	38.69	57.96	2.24	1.11	77.89	16.61	4.33	1.17
CAC-3	35.38	60.68	2.33	1.62	75.13	18.34	4.74	1.8
CAC-4	34.39	59.92	3.14	2.55	72.75	18.04	6.37	2.83

refractive index value. The CAC-3 samples exhibited the lowest refractive index levels. A low surface roughness is a factor in

displaying a low refractive index.<sup>65,76</sup> From the SEM results, the lowest roughness was noted in the CAC-3 sample, which was

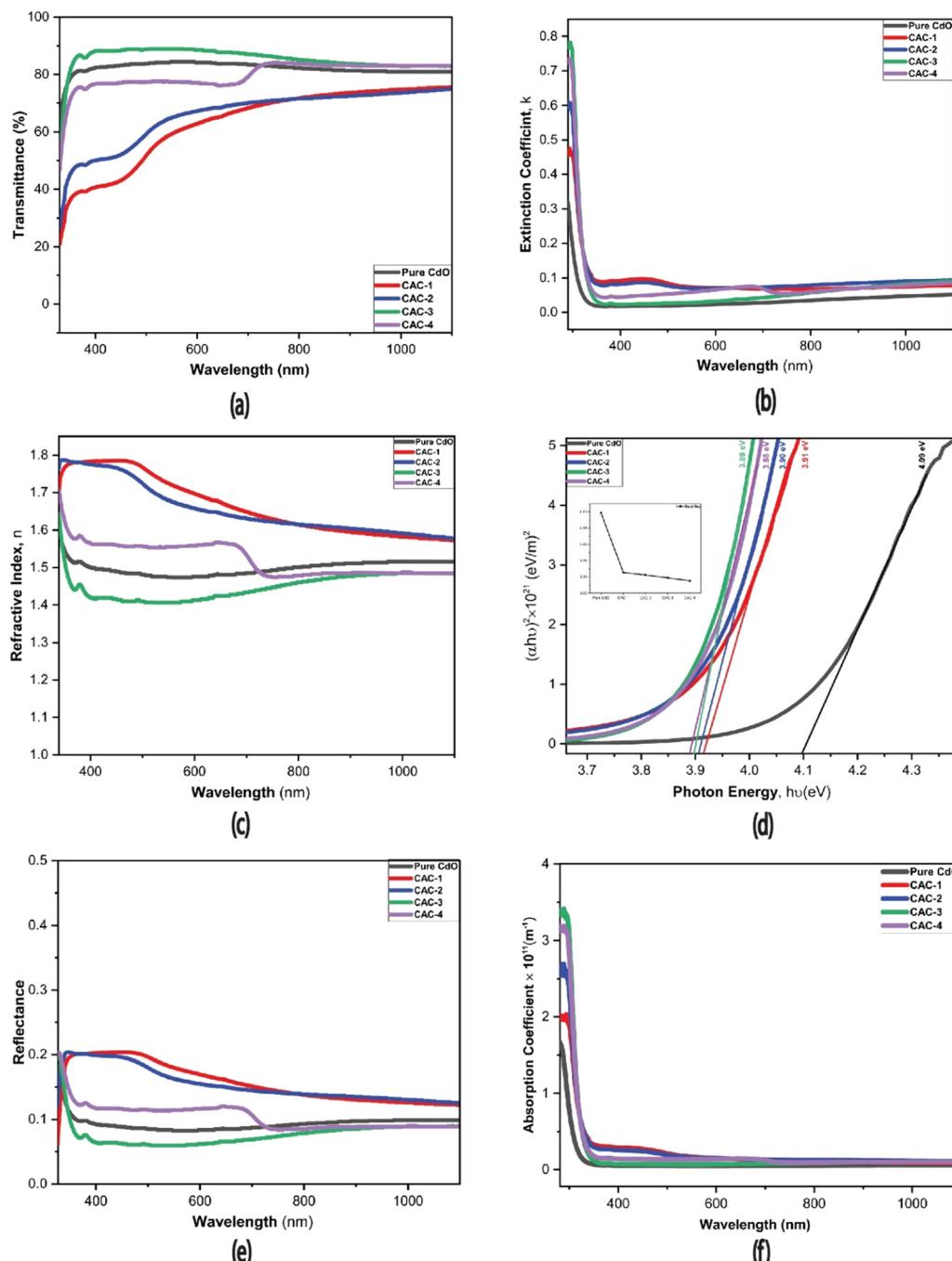


Fig. 6 Variations of the (a) optical transmittance, (b) extinction coefficient, (c) refractive index, (d) optical band gap with photon energy, (e) reflectance, and (f) absorption coefficient of the deposited thin films.

the reason for its lowest refractive index. So, this finding was supported by the SEM results.

The direct band gap of the undoped and co-doped CdO thin films was measured by plotting  $(\alpha h\nu)^2$  vs.  $(h\nu)$ , as shown in Fig. 6(d). A band gap of 4.09 eV was found in the undoped CdO thin film, and a decreased trend was observed with increasing the doping concentration, with bandgaps of 3.91, 3.90, 3.89, and 3.88 eV. Here, we noticed that the double-doped sample maintained a monotonically decreasing series.

Although undoped CdO thin films are recognized for having a low band-gap value,<sup>77</sup> it can be changed by applying different temperatures, deposition techniques, and thicknesses. Radhika *et al.* proposed a higher band gap of 3.8 eV for pure CdO thin films, which aligned with our recent research findings.<sup>78</sup> The creation of a high dislocation density has the potential to enhance the energy band-gap value also.<sup>79</sup> Usharani *et al.*<sup>77</sup> and Azzaoui *et al.*<sup>58</sup> reported band-gap values of 2.46 and 2.41 eV, respectively, for pure CdO thin films, where the



dislocation density was noticed on  $10^{-4}$  (lines per  $\text{nm}^2$ ) scales. In our present work, the dislocation density was found on a scale of  $10^{-3}$  (lines per  $\text{nm}^2$ ), which is greater than in previous work and resulted in the higher band-gap value. The reduced band-gap value of cobalt-doped CdO was caused by the greater sp-d interactions of exchange involving the sp band electrons and the localized d electrons of  $\text{Co}^{2+}$  ions.<sup>80</sup> Dual-doped CdO films,<sup>46,81</sup> Fe-doped CdO NPs<sup>82</sup> and co-doped ZnO NPs<sup>83</sup> were also reported to display a similar behavior. Dakhel *et al.*<sup>84</sup> suggested that the band gap decreases due to the interposition of impurity energy levels into the intrinsic host CdO band gap.

Band-gap broadening or a blue-shift was observed with decreasing the doping concentration, which occurred with a higher carrier concentration,<sup>66</sup> as also supported by the XRD results. This can also be explained by the Burstein-Moss effect.<sup>85,86</sup> Khalaf *et al.*<sup>87</sup> found a similar band-gap range of 3.95–3.18 eV when cobalt was doped in to a CdO thin film. Nithin *et al.*<sup>47</sup> and Sahin *et al.*<sup>81</sup> also elicited that a lower band-gap value occurs with a higher doping concentration. However, Velusamy *et al.*<sup>66</sup> disclosed that a blue-shift of the band-gap value occurred with increasing the doping concentration.

Moreover, factors like the presence of impurities, the deposition conditions, grain size, thickness, and crystallinity can affect the energy band-gap value.<sup>88</sup> Overall, a higher band-gap value is applicable in military, power conversion, and radio applications for allowing adequately handling power of high voltages, temperatures, and frequencies.<sup>21</sup>

Fig. 6(e) and (f) show the reflectance and absorption coefficient values of the undoped and dual-doped CdO thin films. By increasing the doping concentration, the reflectance value increased, albeit for the CAC-3 sample, a lower reflectance was noticed. Higher roughness is the main factor responsible for higher reflectance.<sup>46</sup> The reflectance values were supported by the SEM results, where a lower roughness was found in the CAC-3 sample. Fig. 6(f) shows that the absorption coefficient increased with increasing the doping concentration, but all samples showed a lower absorption coefficient, which was suitable for window-layer applications.<sup>89</sup>

### 3.4 Electrical analysis

Fig. 7(a) illustrates that the doped samples had higher resistivity compared to the pure CdO sample, and the CAC-3 sample showed the highest resistivity, *i.e.*,  $3.32 \times 10^{-2} \Omega \text{ cm}$ .

The ineffectiveness of the dopants in donating free electrons was the factor that increased the resistivity.<sup>25,90,91</sup> A lower Hall mobility and carrier concentration were also noticed in the CAC-3 sample, which are other factors for its highest resistivity.

The decrease in resistivity can be attributed to this higher crystallinity.<sup>21</sup> According to the XRD analysis, the sample with the lowest conductivity (CAC-3) had the lowest crystallinity too (56%), while the sample with the highest conductivity (pure CdO) had the highest crystallinity (85%). The increase in the quantity of defects within the thin films was caused by a decrease in the crystallinity.<sup>21</sup> Compared to pure CdO, this increase in flaws led to greater resistance. Thus, there was a strong correlation between the effect of this on the electrical properties and the XRD results.

Using the Hall effect, other electrical properties such as the Hall mobility ( $\mu_H$ ), carrier concentration ( $n$ ), and Hall coefficient ( $R_H$ ) of the undoped and co-doped CdO thin films were also measured at room temperature, and are summarized in Table 4. The negative sign of the Hall coefficient alludes to the fact that the carriers of all the samples were n-type carriers.

The variations of the Hall mobility ( $\mu_H$ ) and carrier concentration ( $n$ ) are shown schematically in Fig. 7(b). We noticed that with increasing the doping concentration, the carrier concentration ( $n$ ) decreased, and the value of all samples was in the order of  $\sim 10^{19} (\text{cm}^{-3})$ , which matches the reported value for Sr-doped CdO films<sup>64</sup> and Y-doped CdO thin films.<sup>25</sup> The increase in interstitial impurities was the main factor for the decreasing carrier concentration.<sup>25,92</sup> Numerous researchers have examined the Hall mobility of cadmium oxide (CdO) thin films with various metal dopants, such as tin-doped CdO, where the impact of grain boundary scattering has been identified as a significant factor that influences the electron-transport mechanism.<sup>93</sup> In the current work, the reduction in grain boundary scattering was the primary factor contributing

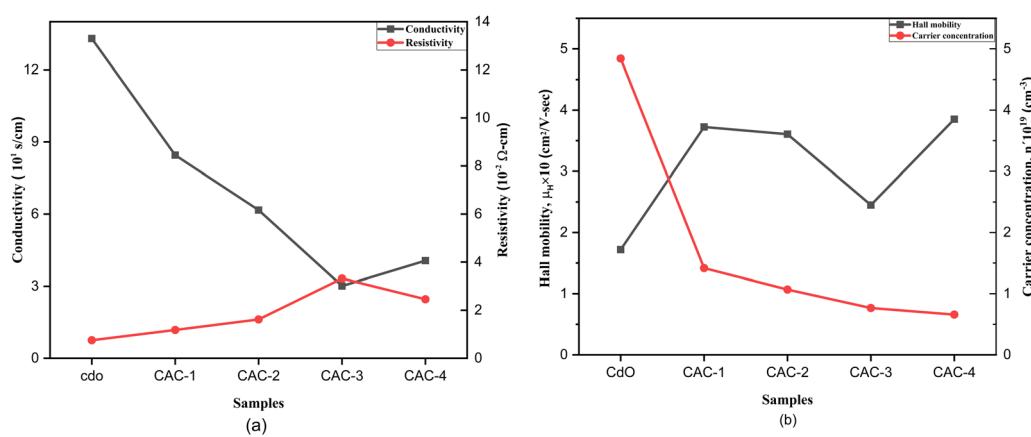


Fig. 7 Variations of the (a) resistivity and conductivity, and (b) Hall mobility ( $\mu_H$ ) and carrier concentrations ( $n$ ) of the pure CdO and Ag-Co dual-doped CdO thin films.



Table 4 Electrical properties of the pure CdO and Ag–Co dual-doped CdO thin films

Sample	Resistivity ( $\Omega$ cm)	Hall mobility, $\mu_H$ ( $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ )	Carrier concentrations, $n$ ( $\text{cm}^{-3}$ )	Hall coefficient, $R_H$ ( $\text{cm}^3 \text{C}^{-1}$ )
CdO	$0.75 \times 10^{-2}$	17.19	$4.84 \times 10^{19}$	-0.12
CAC-1	$1.18 \times 10^{-2}$	37.23	$1.41 \times 10^{19}$	-0.44
CAC-2	$1.62 \times 10^{-2}$	36.05	$1.06 \times 10^{19}$	-0.58
CAC-3	$3.32 \times 10^{-2}$	24.49	$0.76 \times 10^{19}$	-0.81
CAC-4	$2.46 \times 10^{-2}$	38.51	$0.65 \times 10^{19}$	-0.94

Table 5 Comparison of the electrical and optical properties in the present work with some reported literature values

Method	Doping content	Resistivity ( $\Omega$ cm)	Carrier concentration ( $\text{cm}^{-3}$ )	Mobility ( $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ )	% T	Ref.
Spray pyrolysis	3% Ag + 3% Co + 94% CdO	$3.32 \times 10^{-2}$	$0.76 \times 10^{19}$	24.49	89	This work
Spray pyrolysis	100% CdO	$0.75 \times 10^{-2}$	$4.84 \times 10^{19}$	17.19	84	This work
Spray pyrolysis	3% Sr + 97% CdO	$\sim 6.2 \times 10^{-4}$	$1.84 \times 10^{19}$	719	~65	64
Spray pyrolysis	0.25% Co + 99.75% CdO	$7.31 \times 10^{-2}$	$1.11 \times 10^{20}$	80	~83	66
DC sputtering	0.5 wt% Al + 0.5 wt% In	$7.9 \times 10^{-4}$			~80	96
Dual DC sputtering	0.5 wt% $\text{Al}_2\text{O}_3$ doped ZnO + 5 wt% ZnO doped $\text{In}_2\text{O}_3$	$2.32 \times 10^{-4}$	$\sim 10^{21}$	~28	~85	97
RF sputtering	1–7 at% Al + 1–7 at% In	$\sim 2.5 \times 10^{-2}$	$5 \times 10^{19}$	6.3	>80	98
Chemical spray technique	1 at% Al + 3 at% In	$\sim 3.03 \times 10^{-2}$	$\sim 6 \times 10^{20}$	~2	~80	99
Ultrasonic spray pyrolysis	1.5 at% Al + 1.5 at% In	$2.35 \times 10^{-3}$			70	100

to the increase in carrier mobility.<sup>81</sup> Among the doped samples, CAC-3 exhibited the lowest carrier mobility due to the increase in scattering effect. Additionally, for a lower carrier concentration, the overall lowest conductivity was noticed in the CAC-3 sample. With increasing the doping concentration, the Hall mobility increased, and the highest value was  $38.51 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$  found in the CAC-4 sample, which also had the lowest value of carrier concentration. Due to the decreased carrier concentration, the Hall mobility increased.<sup>64</sup> However, Drude's theory is well supported by the behavior of the carrier concentration and Hall mobility in both doped and undoped CdO films.<sup>94,95</sup> Some electrical and optical data comparisons are given in Table 5 for different types of samples deposited in different methods.

### 3.5 Thermal analysis

Fig. 8 shows the thermal conductivity, thermal diffusivity, and specific heat measured at room temperature. It can be seen that

the thermal conductivity and the specific heat increased with increasing the doping percentage from 1% to 4% for the Ag–Co dual-doped elements.

The thermal conductivity is the capacity of a material to conduct heat. The thermal conductivity value is the sum of the lattice thermal conductivity and electronic thermal conductivity. Lattice thermal conductivity dominates the total thermal conductivity. Electron–phonon interactions strongly affect the heat-transfer mechanism of films.<sup>101</sup> The introduction of possible defects and porosity can lead to an increase in phonon scattering, which can reduce the thermal conductivity.<sup>102</sup> Varghese *et al.*<sup>103</sup> highlighted that variations in the defects and porosity could lead to thermal conductivity values in the range of  $1\text{--}3 \text{ W m}^{-1} \text{ K}^{-1}$ , and this present work also found the value of thermal conductivity was in the same range. The lowest thermal conductivity value of the pure CdO sample was  $1.70 \text{ W m}^{-1} \text{ K}^{-1}$ . Lower thermal conductivity has also been found in Cd-rich samples, as observed

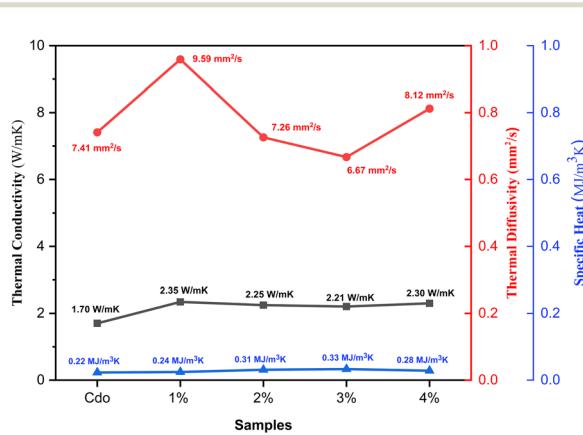


Fig. 8 Plots of the thermal conductivity, thermal diffusivity, and specific heat of the pure CdO and Ag–Co dual-doped CdO thin films.

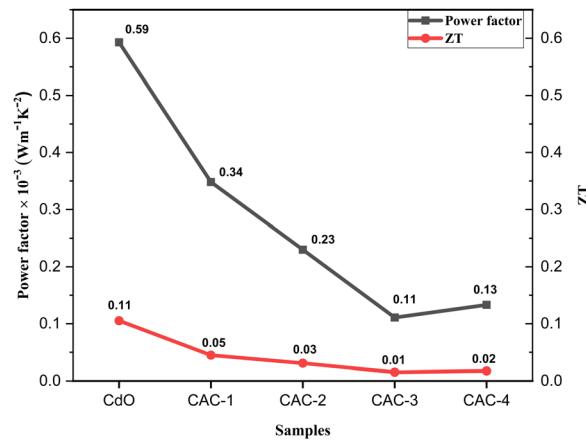


Fig. 9 Power factors and figures of merit (ZT) of the pure CdO and Ag–Co dual-doped CdO thin films.



Table 6 Comparison of the thermoelectric data in this work with some reported literature values

Sample	System	$PF \times 10^{-3} (\text{W m}^{-1} \text{K}^{-2})$	$ZT$	Temperature (K)	Ref.
ZnO	Thin film	0.18	0.017	333	112
ZnO	Thin film	0.11	0.009	300	113
ZnO	Thin film	0.33	0.003	300	114
ZnO/Si	Thin film	0.04	0.004	283	115
ZnO/Al	Thin film	0.08	0.11	300	116
ZnO/Al	Thin film	0.2	0.121	320	117
ZnO(Ga,In)	Thin film	0.3	0.186	573	118
ZnO:Al/silica	Thin film	0.11	0.007	300	119
Sc:ZnCdO	Bulk	0.71	0.30	1173	120
$\text{Cd}_{1-x}\text{Zn}_x\text{O}$	Bulk	1.1	0.52	1000	121
ZnO-CdO	Thin film	0.27	0.59	413	122
CdO	Thin film	0.59	0.11	303	This work
CdO (Ag 3%, Co 3%)	Thin film	0.11	0.01	303	This work

by A. M. M. T. Karim *et al.*<sup>104</sup> In Cd-rich samples, higher electrical resistivity was also noticed. These would be helpful for achieving low thermal conductivity and high electrical performance in thermoelectric devices.<sup>105,106</sup> A moderate range of thermal conductivity is useful in electronics as a thermal interface material, where moderate heat conduction is required to manage heat transfer between heat sinks and components. As the doping concentration increased, the value of the specific heat increased, and the highest specific heat of the CAC-3 sample was  $0.33 \text{ MJ m}^{-3} \text{ K}^{-1}$ , suggesting it would be useful in applications requiring moderate heat storage, temperature stability, and energy efficiency.<sup>107</sup> Thermal diffusivity is a measurement of the heat conductivity through a material relative to its ability to store thermal energy.<sup>108</sup> The highest thermal diffusivity of  $9.59 \text{ mm}^2 \text{ s}^{-1}$  was noticed in the CAC-1 sample, but the overall value was lower, and this would be helpful in applications where the material needs to maintain a stable temperature or provide thermal buffering (Fig. 9).

The power factor (PF) of the pure CdO and Ag–Co dual-doped CdO thin films was calculated from eqn (17). The dimensionless quantity used to determine and evaluate the performance of a thermoelectric material is called the figure of merit, which is given by eqn (18).

The power factor of all the samples decreased with increasing the doping concentration, and the lowest value of  $0.11 \times 10^{-3} \text{ W m}^{-1} \text{ K}^{-2}$  was noticed in the CAC-3 sample. The highest power factor was  $0.59 \times 10^{-3} \text{ W m}^{-1} \text{ K}^{-2}$  for the pure CdO thin film. The figure of merit in a thermoelectric device reflects its effectiveness in converting heat to electricity or thermoelectric performance. By increasing the doping concentration, the  $ZT$  value decreased. The highest  $ZT$  value noticed in pure CdO was 0.11, and the lowest was in CAC-3, which was 0.01.

According to the formula, materials with a high thermal conductivity, Seebeck coefficient, and low thermal conductivity are considered suitable thermoelectric materials.<sup>109,110</sup> The highest  $ZT$  value is applicable in thermoelectric devices (Table 6).<sup>111</sup>

## 4 Conclusion

In this study, pure CdO and Ag–Co dual-doped CdO thin films were deposited on a glass substrate by using the spray-pyrolysis

method at 400 °C. The significant structural, morphological, optoelectrical and thermal properties were explored for offering innovative solutions across various fields, including electronics, photovoltaics, and solar cell applications. The slight rightward peak position shift noted in the XRD patterns confirmed the successful Ag and Co doping into the CdO lattice. Scanning electron microscopy analysis illustrated that all the films had spherical-shaped agglomerated grains, and, the 3% Ag–Co doped CdO film exhibited a smooth surface compared to the other films, which indicates it can be used in gas sensors and optoelectronics applications, as a smooth surface would minimize scattering losses. The maximum optical transmittance of 89% from this study can be essential for applications in solar cells, where light absorption is critical for energy conversion. Hall measurements were used to quantify the lowest resistance, which was  $1.18 \times 10^{-2} \text{ Ohm cm}$ , among the doped samples, which would be suitable for applications requiring efficient charge transport. Thermal analysis indicated that the thermal conductivity increased with increasing the doping concentration, which could be useful in electronic sensors and thermoelectric applications. This study provides new insights that could be useful for the advancements of optoelectronics applications.

## Author contributions

Ishraque Karim: writing – review & editing, writing – original draft, funding acquisition, methodology, investigation, formal analysis, data curation, conceptualization. M Ashikul Haque Naeem: writing – review & editing, writing – original draft, methodology, investigation, formal analysis, data curation, conceptualization. Ahmed Sidrat Rahman Ayon: writing – editing. Md. Abdus Sattar: writing – review & editing, supervision, project administration, formal analysis, conceptualization. Md. Abdus Sabur: characterization, review, project administration, supervision. Aninda Nafis Ahmed: resource & supervision.

## Data availability

Data will be made available on request.



## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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