





Cite this: *Phys. Chem. Chem. Phys.*,
2025, 27, 9275

Metal intercalation induced magnetic modulation in VS₂ bilayers: a first principles study†

Dantong Li, Xiaocheng Zhou,* Yu Wang  and Yafei Li *

Two-dimensional (2D) magnetic materials have emerged as highly promising candidates for spintronic applications owing to their unique properties. However, they tend to exhibit antiferromagnetic coupling when stacked, which limits their broader applications. It is of critical importance to manipulate the magnetic coupling and explore their applications in low-energy-consumption spintronic devices. Here, we theoretically investigated the electronic and magnetic properties of VS₂ bilayers intercalated with transition metals (TMs) using density functional theory (DFT) computations and the nonequilibrium Green's function (NEGF) method. It is revealed that metal intercalation can significantly enhance the ferromagnetic exchange interaction. The TM-intercalated VS₂ bilayers (TM-VS₂) exhibit diverse electronic and magnetic properties, which can be precisely tuned *via* controlling the type and concentration of intercalated metals. The spin-polarized transport calculations demonstrate that the TM-VS₂ bilayer with half-metallicity exhibits pronounced spin filtering properties. Our theoretical study provides a promising route to design and modulate the magnetic properties of 2D ferromagnets for their applications in advanced spintronic devices.

Received 5th March 2025,
Accepted 31st March 2025

DOI: 10.1039/d5cp00869g

rsc.li/pccp

1. Introduction

Two-dimensional (2D) materials have attracted great attention due to their novel properties and potential applications. Their unique quantum confinement effect helps them effectively overcome the short-channel effect in miniaturized electronic devices.^{1–3} Among them, 2D magnetic materials have attracted extensive attention and demonstrated promising applications beyond conventional semiconductors with the introduction of spin freedom.^{4–6} The spin state of electrons can be used for information generation, storage, and transmission, which is essential for low power spintronic technologies. Ideally, spintronic devices can generate pure spin currents, substantially reducing Joule heating and achieving ultra-high circuit integration density.⁷ However, most of the 2D materials do not have inherent magnetism, which limits their widespread application in spintronics. Previously, several approaches were used to successfully introduce magnetism into 2D materials, such as atomic doping, defect engineering, and edge effects.^{8–10} However, the induced magnetism is often highly localized and weak, bringing challenges for their widespread application in practical devices.¹¹ Therefore, it is imperative to discover 2D

ferromagnets with strong magnetic anisotropy and ordered long-range magnetic exchange interaction. In recent years, several 2D magnetic materials were obtained experimentally, such as ferromagnetic (FM) CrI₃,¹² Cr₂Ge₂Te₆,¹³ Fe₃GeTe₂,¹⁴ MnSe₂,⁶ and VX₂^{15,16} and antiferromagnetic (AFM) FePS₃¹⁷ and NiPS₃.¹⁸

In two-dimensional magnetic materials, interlayer magnetic coupling plays a crucial role in determining magnetic properties. For instance, the CrI₃ monolayer exhibits intrinsic intralayer ferromagnetism. When two layers of CrI₃ are stacked, they tend to exhibit interlayer antiferromagnetic coupling in monoclinic stacking and ferromagnetic coupling in rhombohedral stacking.¹⁹ It has been revealed that the stacking-dependent magnetism originates from the interlayer orbital hybridization and charge accumulation. Still, most of the 2D magnets possess weak coercive fields due to the relatively weak magnetocrystalline anisotropy. Consequently, it is necessary to explore effective strategies to enhance and tune the magnetism of 2D magnets. Intercalation of atoms into the van der Waals (vdW) gap is a feasible approach for tuning and enhancing magnetism in 2D materials due to the induced interfacial charge redistribution and long-range spin-exchange interaction. So far, various atoms have been intercalated into 2D materials with controlled concentration and order. For example, Li intercalation in layered Fe₃GeTe₂ remarkably enhances the ferromagnetism and increases the ferromagnetic transition temperature.²⁰ The intercalation of Fe and Cr into NbS₂ and TaS₂ leads to distinct magnetic behaviors, depending on the

Key Laboratory of Numerical Simulation for Large Scale Complex Systems (Ministry of Education of PRC), School of Chemistry and Materials Science, Nanjing Normal University, Nanjing 210023, China. E-mail: xczhou@njnu.edu.cn, liyafei@njnu.edu.cn

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d5cp00869g>

type of intercalant and the intercalation concentration.²¹ In addition, self-intercalation was also successfully achieved in a broad class of 2D materials, such as layered materials (TMDs), which provides a powerful method for tuning and controlling magnetism.²²

As a typical member of transition metal dichalcogenides, VS₂ has two possible phases: octahedral coordinated (T-phase) and trigonal prismatic (H-phase).¹⁵ They have quite different electronic and magnetic properties. The T-phase VS₂ monolayer has been widely studied due to its high conductivity, excellent electrocatalytic activity, and ferromagnetism. However, the H-phase VS₂ is a ferromagnetic semiconductor with a large magnetic moment.²³ Notably, when two layers of H-phase VS₂ are stacked together, they show antiferromagnetic interlayer coupling with a zero net magnetic moment.²⁴ Although antiferromagnetic spintronics has been successfully used in various fields,²⁵ effective regulation of magnetic coupling is still desired for satisfying different requirements of spintronic devices.

Here, we performed density functional theory (DFT) and non-equilibrium Green's function (NEGF) calculations to study the electronic and magnetic properties of transition metal intercalated H-phase VS₂ bilayers (TM-VS₂) for their applications in spintronic nanodevices. The geometries, ground state energies, band structures, and magnetic behaviors of VS₂ bilayers with different intercalation configurations, metals and concentrations have been investigated. The results demonstrated that the VS₂ bilayers exhibit a rich variety of magnetic properties depending on the type of intercalation. Notably, the self-intercalated V-VS₂ bilayer exhibits the half-metallic characteristic, which is crucial for achieving spin filtering effects. Our results indicate that metal intercalation offers a powerful approach to design 2D spintronic devices with tunable magnetic functionalities.

2. Computational methods

Our calculations are based on density functional theory (DFT), implemented using the Vienna *ab initio* simulation package (VASP).²⁶ The exchange–correlation interaction is described using the Perdew–Burke–Ernzerhof (PBE) functional²⁷ based on the generalized gradient approximation (GGA).²⁸ To avoid interactions between adjacent layers, a vacuum of 15 Å along the *z*-direction is used. The energy cut-off is set to be 450 eV and the convergence criterion for energy is 10^{−5} eV. Geometry optimization is performed until the force acting on each atom is less than 0.02 eV Å^{−1}. The Brillouin zone was sampled with a 5 × 5 × 1 *k*-mesh grid. *Ab initio* molecular dynamics (AIMD) simulations were performed for 10 ps at 500 K with a time step of 1 fs. The temperature was controlled using the Nosé–Hoover method.²⁹ To appropriately describe the strongly correlated electrons in *d* orbitals of TM atoms, we use the PBE+*U* method³⁰ with the effective *U* values of 3.0 eV for V, Co, and Mn atoms and 2.0 eV for the Cr atom.^{5,23,31–33}

We further designed the device based on TM-VS₂ to investigate its potential applications in spintronics. The spin-polarized transport simulations were implemented *via* the NANODCAL package,³⁴ which combines DFT with the non-equilibrium Green's function. A cutoff energy of 3000 eV is adopted and the convergence criterion for both the Hamiltonian and the density matrix is set to 10^{−4} eV. The *k*-point grid meshes are 1 × 3 × 1 and 100 × 3 × 1 for the center region and the electrode, respectively. The Landauer–Buttiker formula is used to calculate the probability of incoming electrons transferred from one electrode to another with the specific energy (*E*) as follows:

$$T(E, V_b) = \frac{e^2}{h} \text{Tr} \left[\Gamma_L(E) G_C(E) \Gamma_R(E) G_C^\dagger(E) \right] \quad (1)$$

where $\Gamma_{L/R}(E)$ is the coupling matrix of the L/R electrode and $G_C(E)$ and $G_C^\dagger(E)$ are the retarded and advanced Green's functions, respectively.^{35,36}

3. Results and discussion

3.1. Structure and stability of TM-VS₂

To investigate the properties of VS₂ with intercalated transition metals, we started from the atomic structures of the pristine VS₂ bilayer with different stacking geometries. The most stable stacking geometry is shown in Fig. 1a. The relaxed lattice constants of the 2√3 × 2√3 × 1 supercell are *a* = *b* = 11.02 Å, and the corresponding V–S bond length and the distance between two VS₂ layers are 2.38 Å and 3.26 Å, respectively. Our calculations show that the VS₂ monolayer is ferromagnetic while the VS₂ bilayer prefers antiferromagnetic interlayer coupling (which is energetically lower than the FM interlayer

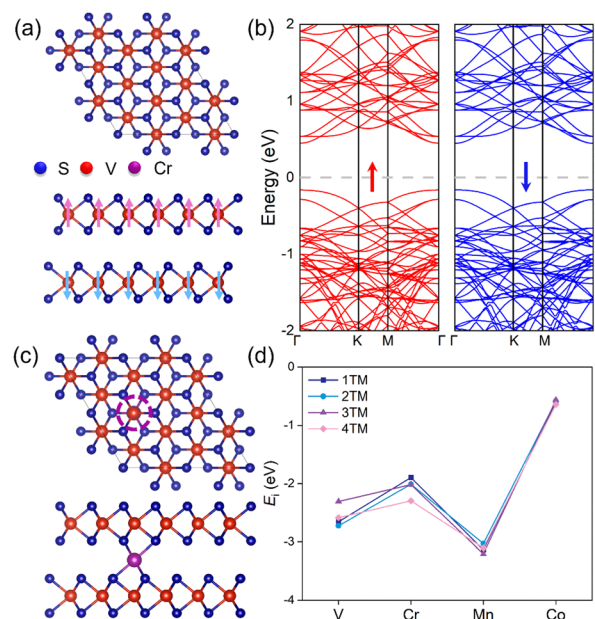


Fig. 1 (a) Top and side views of the pristine VS₂ bilayer. (b) Spin-polarized band structures of the VS₂ bilayer. The Fermi level is set to zero. (c) Top and side views of the 1TM-VS₂ bilayer. (d) The intercalation energies of the TM-VS₂ bilayer with different metals and concentrations.

coupling by 26 meV per supercell), agreeing well with previous studies.²⁴ The calculated electronic properties presented in Fig. 1b indicate that the AFM VS₂ bilayer is semiconducting with a direct bandgap of 0.61 eV.

Owing to the layered structure, intercalation of transition metal atoms into the bilayer gap is experimentally feasible.³⁷ It has emerged as an effective method to modulate the properties of 2D materials. To systematically explore the effect of intercalated atoms on magnetic coupling, we considered possible intercalation sites for transition metals in the VS₂ bilayer and fully relaxed the structures to identify the optimal intercalation position (Fig. S1, ESI†). It turned out that the most energetically favorable site for an intercalant is at the top of the V atom, where the intercalated atom bonds with six S atoms to form an octahedral coordination in the vdW gap (Fig. 1c). This site preference may be attributed to the geometric characteristics of octahedral coordination, which provides a more favorable six-fold coordination environment and requires minimal structural reorganization. Compared with the pristine bilayer structure, the lattice constants of the intercalated bilayer are almost unchanged, while the interlayer distances are decreased from 3.26 Å to 3.11 Å, indicating the enhanced interactions within the bilayer. To evaluate the energetic stability of intercalated atoms, we define the intercalation energy as follows:

$$E_i = (E_{\text{tot}} - E_{\text{bilayer}} - nE_{\text{TM}})/n \quad (2)$$

where E_{tot} and E_{bilayer} are the energies of bilayer VS₂ with and without intercalation, respectively; E_{TM} is the energy of a single TM atom and n denotes the number of intercalated atoms. As shown in Fig. 1d, the intercalation energies are in the range from -0.56 to -3.21 eV, suggesting that the intercalation processes are all exothermic and thermodynamically stable. In addition, we also performed 10 ps AIMD simulations at 500 K for the high concentration (33%) intercalation system (Fig. S3, ESI†). The results demonstrate that the structure maintains well-preserved integrity with no observable structural distortion or phase transition. We intercalated various metal atoms (TM = V, Cr, Mn, and Co) of varying concentrations into the interlayer space. The intercalation concentration is defined as the percentage of initial octahedral vacancy sites occupied by intercalated TM atoms. In our system, one, two, three, and four intercalated atoms correspond to intercalation concentrations of 8.3%, 16.7%, 25%, and 33.3%, respectively. High concentration intercalation has been experimentally achieved in several systems; therefore, the intercalation concentration considered in our study not only preserves structural robustness but also is experimentally feasible.^{38,39}

3.2. Magnetic properties of TM-VS₂

After intercalation, the intralayer ferromagnetic coupling remains robust, while the interlayer magnetic coupling of the VS₂ bilayer will be affected. Importantly, the intercalated TM atoms also exhibit various magnetic states, which will decisively influence the overall properties of the systems. To figure out the magnetic ground states of the systems, we calculated the energy difference defined as follows:

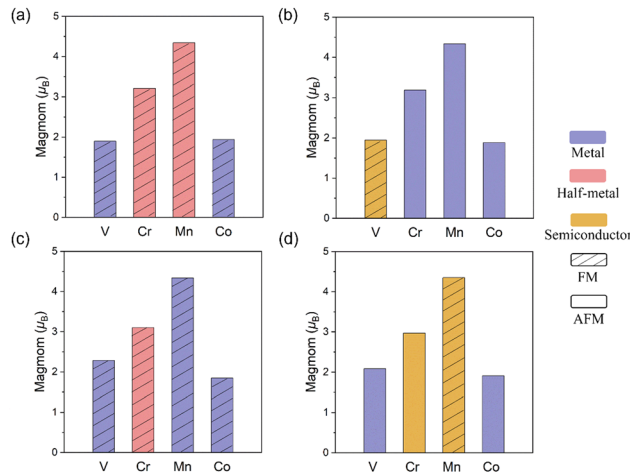


Fig. 2 A summary of the magnetic properties of TM-VS₂ bilayers with the intercalation concentrations of (a) 8.3%, (b) 16.7%, (c) 25%, and (d) 33.3%.

$$E = E_{\text{AFM}} - E_{\text{FM}} \quad (3)$$

where E_{AFM} and E_{FM} represent the energies of the systems with FM and AFM spin configurations for intercalated atoms, respectively. Here, we have considered both the FM and AFM interlayer coupling states of bilayer VS₂. The energy difference calculations were performed based on the energetically more favorable interlayer magnetic coupling states. The results are shown in Fig. 2. When transition metal atoms are intercalated into the vdW gap, the interlayer coupling remains antiferromagnetic for V, Cr, and Co and becomes ferromagnetic for Mn. This result is consistent with our calculated E_i . As shown in Fig. 1d, the values of E_i for the Mn-intercalated systems are more negative, indicating the stronger interaction between the intercalated atoms and the pristine VS₂ bilayer. The magnetic exchange variation can be understood as a result of two competing magnetic coupling mechanisms. The V (+4) in pristine VS₂ adopts a trigonal-prismatic structure bonding to six S atoms. The d orbitals of V are divided into three groups, $a'_1(d_{z^2})$, $e'(d_{xy}, d_{x^2-y^2})$, and $e''(d_{xz}, d_{yz})$, and the a'_1 orbital is the lowest level. The single electron in the 3d¹ configuration occupies one of the spin-polarized d_{z^2} orbitals. This is consistent with our calculations that each V atom exhibits a magnetic moment of about 1 μ_B . The interlayer AFM coupling of the VS₂ bilayer is facilitated by the direct V-V exchange interaction. The d orbitals of intercalated transition atoms are divided into two groups, $t_{2g}(d_{xy}, d_{xz}, d_{yz})$ and $e_g(d_{z^2}, d_{x^2-y^2})$. The interlayer magnetic coupling almost remains antiferromagnetic for V, Cr, and Co intercalation but becomes ferromagnetic for Mn intercalation. This difference arises primarily from the distinct d-electron configurations of the metals, which influence their interactions with the V atoms in the pristine bilayer. The eventual interlayer magnetic coupling is a consequence of the competition between direct and indirect exchange interactions. As shown in Fig. 2, the calculated local magnetic moments on each intercalated V, Cr, Mn, and Co atoms are approximately 2, 3, 4, and 2 μ_B , respectively. The spin configurations of one and

three intercalated atoms are obviously ferromagnetic, whereas those of two and four intercalated atoms depend on the type and concentration of the metals. In the case of two intercalated atoms, we found that only 2V tend to adopt the FM configuration, while 2Cr, 2Mn, and 2Co prefer the AFM configuration. When four atoms are intercalated, 4V, 4Cr, and 4Co favor the AFM configuration, and only 4Mn exhibits the FM configuration. Considering diverse spin configurations of intercalated atoms, our results demonstrate that TM-VS₂ systems exhibit a rich variety of properties, including ferromagnetic metals, ferromagnetic half-metals, ferromagnetic semiconductors, antiferromagnetic metals, and antiferromagnetic semiconductors.

3.3 Electronic properties of TM-VS₂

Considering the diverse properties of VS₂ systems induced by intercalated TM atoms, we take Cr-VS₂ as a representative example to conduct the in-depth analysis. In Fig. 3a, the spin-polarized band structures of 1Cr-VS₂ reveal that the system is converted into half-metallic, suggesting its potential to achieve the spin filtering effect. As observed from the band structures, intercalation shifts the Fermi level upwards, resulting in the bands crossing the E_F in the spin-up channel while retaining the semiconducting feature in the spin-down channel. The spin density shown in Fig. 3b indicates that the V atoms in the bilayer favor FM intralayer coupling and AFM interlayer coupling. The asymmetric spin-up and spin-down projected densities of states (PDOS) confirm the ferromagnetic ground state of 1Cr-VS₂. The localized state around the Fermi level is primarily contributed by the Cr_d orbitals, which contributes to the flat bands. It can be observed that there is a significant overlap between the Cr_d and V_d orbitals around the Fermi

level, indicating the strong orbital hybridization between them. This is consistent with our charge density difference (Fig. S5a, ESI†) and Bader charge analysis calculations, which show that about 1.2 electrons are transferred from Cr to the VS₂ bilayer and lead to an electron doping effect. These interactions are primarily contributed by the d_{xz} and d_{yz} orbitals of the Cr atom. The above results demonstrated that the intercalation of Cr can enhance the hybridization between Cr_d and V_d orbitals and induce the spin redistribution, thereby achieving effective regulation of magnetic properties.

To further investigate the effect of intercalation concentration on magnetic coupling, we computed the band structures and PDOS of Cr-VS₂ at the concentrations of 16.7%, 25%, and 33.3%. When two Cr atoms (16.7%) are intercalated into the vdW gap, they tend to form an AFM spin configuration. As shown in Fig. 4a, the V atoms in the bilayer maintain AFM interlayer coupling, and the two Cr atoms between layers present spin densities in opposite directions. The 2Cr-VS₂ exhibits metallic behavior, which is primarily contributed by the Cr_d orbitals near the Fermi level. As the intercalation concentration increases to 25% and 33.3%, the intercalated Cr atoms induce FM half-metallic and AFM semiconducting characteristics, respectively. The spin densities presented in Fig. 4b and c indicate that three Cr atoms in the bilayer gap align with the same spin orientation, whereas four Cr atoms adopt an AFM spin configuration. Notably, although the VS₂ bilayer exhibits AFM interlayer coupling, the spin configurations of the intercalated Cr atoms significantly influence the magnetic states and moments of V atoms. As seen from the PDOS, 1Cr and 3Cr intercalations produce similar electronic states of VS₂. It is obviously that the enhanced V_d density of states appear in the majority spin channel near the Fermi level due to the FM state of Cr in the interlayer gap. Conversely, 2Cr and 4Cr intercalations generate a distinct type of electronic states of VS₂, where the AFM Cr intercalations influence the electronic states of VS₂ primarily by introducing two spin-polarized peaks

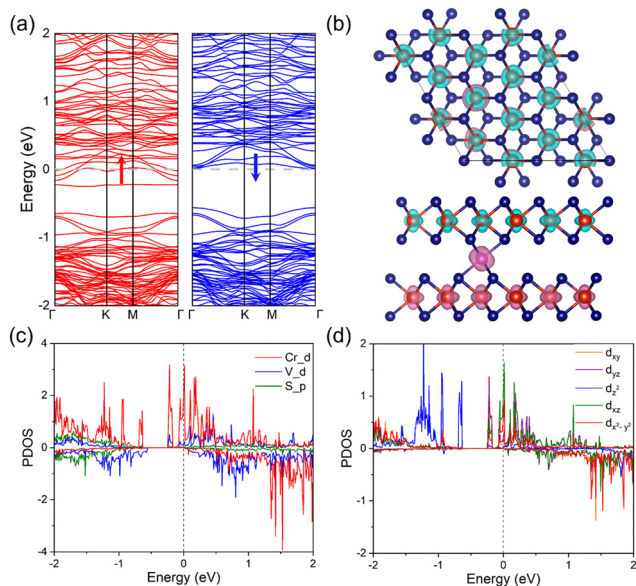


Fig. 3 (a) The spin-polarized band structure. (b) Top and side views of spin density. (c) PDOS of Cr_d, V_d, and S_p orbitals. (d) PDOS of the distinct Cr_d orbitals for Cr-VS₂ at an intercalation concentration of 8.3%. The green and pink colors represent spin-up and spin-down configurations, respectively. The isosurface value is set at $0.02 \text{ e}^- \text{ \AA}^{-3}$.

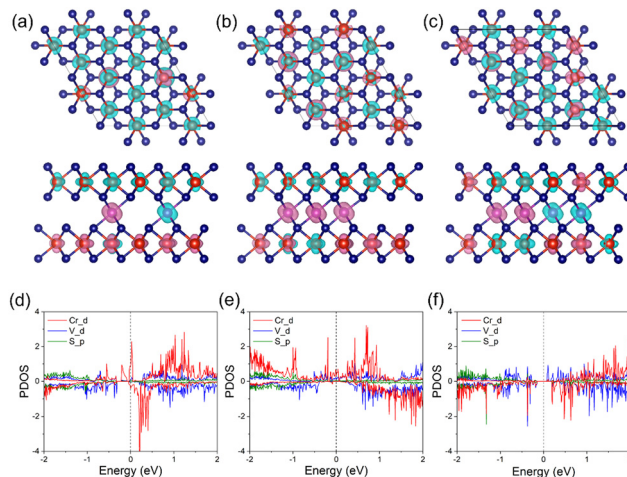


Fig. 4 (a)–(c) Top and side views of spin density and (d)–(f) the spin-polarized PDOS of the Cr_d, V_d, and S_p orbitals of Cr-VS₂ systems at the intercalation concentrations of 16.7%, 25%, and 33.3%, respectively.

below the Fermi level. Particularly, when the doping concentration reaches 33%, the magnetic states of the V atoms adjacent to the intercalated atoms are significantly affected. The four V atoms adjacent to Cr exhibit spin configurations opposite to those of the intercalated atoms, driving the system into the lowest energy state. These findings confirm the significant impact of intercalation concentration and the spin configurations of Cr atoms on the electronic and magnetic properties of Cr-VS₂ systems.

The type of intercalated metal is another critical factor in influencing the magnetic properties of the TM-VS₂ systems. For V and Co intercalations, the TM-VS₂ systems exhibit AFM interlayer coupling between VS₂ layers. Consequently, the PDOS of V-VS₂ and Co-VS₂ are similar to that of Cr-VS₂, with electronic states near the Fermi level predominantly contributed by the TM_d orbitals (Fig. S6 and S8, ESI†). The overlap between TM_d and V_d orbitals indicates significant orbital hybridization. The electronic and magnetic properties of the systems are ultimately determined by the type and concentration of the intercalated atoms. Analysis of the PDOS and spin density reveals that for systems with 1–3 V atoms intercalated, the intercalated V atoms exhibit FM coupling. Among them, 1V-VS₂ and 3V-VS₂ systems demonstrate FM metallic behavior, while the 2V-VS₂ system exhibits the FM semiconducting characteristic, with the conduction band minimum and valence band maximum primarily contributed by the intercalated atoms. When four V atoms are intercalated, they tend to adopt an antiferromagnetic configuration. The combined effects of the intercalated and intrinsic V atoms drive the 4V-VS₂ system toward metallic behavior. The 1Co- and 3Co-VS₂ systems adopt FM configurations, whereas 2Co- and 4Co-VS₂ systems favor AFM configurations. Regardless of the concentration, all Co-VS₂ systems consistently display metallic behaviors, driven by the strong orbital hybridization between Co_d and V_d orbitals near the Fermi level. In contrast, Mn atoms possess unique half-filled d-orbitals, which may alter the spin state of the V atoms and activate direct exchange interactions mediated by the V_d orbitals. This can be corroborated by the PDOS of the Mn-VS₂ system (Fig. S7, ESI†), where the PDOS near the Fermi level is dominated by V_d orbitals, while the contributions from Mn_d orbitals are at deeper energy levels. Consequently, the VS₂ bilayer exhibits ferromagnetic interlayer coupling in the Mn-intercalated system. Taken together, we suggest that intercalation of transition metal atoms with different types and concentrations can significantly modulate the electronic and magnetic properties of VS₂ bilayers, providing theoretical insights for designing functional 2D materials with diverse properties.

3.4. Spin transport properties of the V self-intercalated bilayer

The above results demonstrate that tuning the type and concentration of intercalated transition metal atoms can yield TM-VS₂ systems with diverse electronic properties and magnetic states, which makes them highly promising candidates for various spintronic applications. The intercalated atoms create chemical binding with the intrinsic bilayer, which could facilitate the

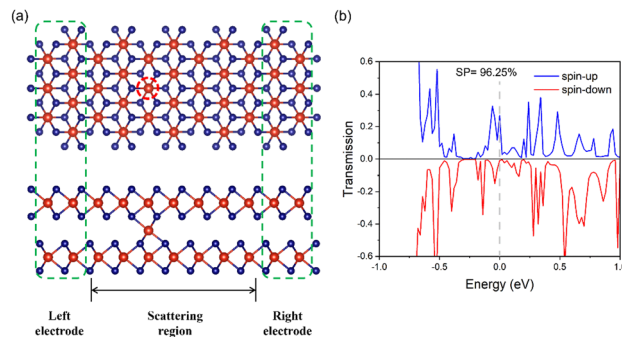


Fig. 5 (a) Schematic diagram of the dual-probe transport device for spin transport calculations, including the semi-infinite left and right electrodes (in dashed) and the central device region. (b) Spin-polarized transmission spectrum of the device at a bias voltage of 0.2 V.

carrier transfer between layers and effectively regulate the spin transport. Notably, self-intercalation can be achieved by controlling the metal chemical potential during the material growth process,³⁹ avoiding the harsh reaction conditions typically required for foreign atom intercalation.

To investigate the spin transport properties of the system, we constructed a dual-probe system for calculations using non-equilibrium Green's function method. As illustrated in Fig. 5a, the dual-probe model consists of three regions: semi-infinite left and right electrode regions composed of pristine bilayer VS₂ and a central device region formed by V-VS₂ (5.6%). The computed spin-polarized transmission spectrum reveals a pronounced spin polarization property. Specifically, at the Fermi level, the transmission spectrum exhibits a pronounced peak in the spin-up channel, while in the spin-down channel it is negligible. The transmission coefficient at the Fermi level reflects the probability of electron transfer through the conductance channel. To evaluate whether the device can be applied in spin filters, the spin polarization (SP) is defined as $SP = (T_{up} - T_{down}) / (T_{up} + T_{down})$, where T_{up} and T_{down} are the spin-up transmission and spin-down transmission at the Fermi level, respectively.⁴⁰ We found that the SP of the device can reach up to 96% at a bias voltage of 0.2 V, indicating its potential application as a spin filter by facilitating current through the spin-up channel.

4. Conclusions

In summary, we have systematically investigated the strategy of modulating the electronic and magnetic properties of the VS₂ bilayer *via* metal intercalation based on the combination of DFT and NEGF calculations. Our results reveal that the interlayer coupling in bilayer VS₂ is affected by the type of intercalated metal; intercalation of V, Cr, and Co preserves the AFM interlayer coupling, whereas Mn intercalation induces a transition to FM interlayer coupling. Moreover, the magnetic characteristics of the TM-VS₂ systems are determined by both the type and concentration of the intercalated atom. Adjusting these parameters could effectively modulate the charge transfer between the intercalants and the VS₂ bilayer, thereby altering

orbital occupation and magnetic exchange interactions, which enables diverse tunable electronic and magnetic properties, spanning from ferromagnetic to antiferromagnetic, as well as metallic, half-metallic, and semiconducting behaviors. Furthermore, our spin transport calculations confirm that the half-metallic system exhibits obvious spin-polarized transmission spectra and spin filtering capability. This study deepens our understanding of the role of metal intercalation in magnetic modulation and provides theoretical guidance for the tailored design of spintronic devices with specific functionalities.

Author contributions

This manuscript was written through contributions from all authors. All authors have given approval to the final version.

Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

The authors are grateful for funding support from the Natural Science Foundation of China (No. 12202209, No. 22173048, and 22425302), and the Natural Science Foundation of Jiangsu Province (BK20220379).

References

- 1 N. Tombros, C. Jozsa, M. Popinciuc, H. T. Jonkman and B. J. V. Wees, *Nature*, 2007, **448**, 571–574.
- 2 E. C. Ahn, *npj 2D Mater. Appl.*, 2020, **4**, 17.
- 3 A. L. Friedman, O. M. J. V. Erve, C. H. Li, J. T. Robinson and B. T. Jonker, *Nat. Commun.*, 2014, **5**, 3161.
- 4 A. Avsar, J. Y. Tan, M. Kurpas, M. Gmitra, K. Watanabe, T. Taniguchi, J. Fabian and B. Özyilmaz, *Nat. Phys.*, 2017, **13**, 888–893.
- 5 M. Bonilla, S. Kolekar, Y. Ma, H. C. Diaz, V. Kalappattil, R. Das, T. Eggers, H. R. Gutierrez, M.-H. Phan and M. Batzill, *Nat. Nanotechnol.*, 2018, **13**, 289–293.
- 6 D. J. O'Hara, T. Zhu, A. H. Trout, A. S. Ahmed, Y. K. Luo, C. H. Lee, M. R. Brenner, S. Rajan, J. A. Gupta, D. W. McComb and R. K. Kawakami, *Nano Lett.*, 2018, **18**, 3125–3131.
- 7 Y. Hu, S. Liu, J. Huang, X. Li and Q. Li, *Nano Lett.*, 2023, **23**, 7890–7896.
- 8 K. Zhang, S. Feng, J. Wang, A. Azcatl, N. Lu, R. Addou, N. Wang, C. Zhou, J. Lerach, V. Bojan, M. J. Kim, L.-Q. Chen, R. M. Wallace, M. Terrones, J. Zhu and J. A. Robinson, *Nano Lett.*, 2015, **15**, 6586–6591.
- 9 O. V. Yazyev, *Phys. Rev. Lett.*, 2008, **101**, 037203.
- 10 Y.-W. Son, M. L. Cohen and S. G. Louie, *Nature*, 2006, **444**, 347–349.
- 11 R. R. Nair, M. Sepioni, I.-L. Tsai, O. Lehtinen, J. Keinonen, A. V. Krasheninnikov, T. Thomson, A. K. Geim and I. V. Grigorieva, *Nat. Phys.*, 2012, **8**, 199–202.
- 12 B. Huang, G. Clark, E. Navarro-Moratalla, D. R. Klein, R. Cheng, K. L. Seyler, D. Zhong, E. Schmidgall, M. A. McGuire, D. H. Cobden, W. Yao, D. Xiao, P. Jarillo-Herrero and X. Xu, *Nature*, 2017, **546**, 270–273.
- 13 C. Gong, L. Li, Z. Li, H. Ji, A. Stern, Y. Xia, T. Cao, W. Bao, C. Wang, Y. Wang, Z. Q. Qiu, R. J. Cava, S. G. Louie, J. Xia and X. Zhang, *Nature*, 2017, **546**, 265–269.
- 14 Y. Deng, Y. Yu, Y. Song, J. Zhang, N. Z. Wang, Z. Sun, Y. Yi, Y. Z. Wu, S. Wu, J. Zhu, J. Wang, X. H. Chen and Y. Zhang, *Nature*, 2018, **563**, 94–99.
- 15 H. Zhang, L.-M. Liu and W.-M. Lau, *J. Mater. Chem. A*, 2013, **1**, 10821.
- 16 M. Bonilla, S. Kolekar, Y. Ma, H. C. Diaz, V. Kalappattil, R. Das, T. Eggers, H. R. Gutierrez, M.-H. Phan and M. Batzill, *Nat. Nanotechnol.*, 2018, **13**, 289–293.
- 17 J.-U. Lee, S. Lee, J. H. Ryoo, S. Kang, T. Y. Kim, P. Kim, C.-H. Park, J.-G. Park and H. Cheong, *Nano Lett.*, 2016, **16**, 7433–7438.
- 18 K. Kim, S. Y. Lim, J.-U. Lee, S. Lee, T. Y. Kim, K. Park, G. S. Jeon, C.-H. Park, J.-G. Park and H. Cheong, *Nat. Commun.*, 2019, **10**, 345.
- 19 B. Niu, T. Su, B. A. Francisco, F. Kargar, X. Huang, M. Lohmann, J. Li, Y. Xu, T. Taniguchi, K. Watanabe, D. Wu, A. Balandin, J. Shi and Y.-T. Cui, *Nano Lett.*, 2020, **20**, 553–558.
- 20 X. Huang, J. Xu, R. Zeng, Q. Jiang, X. Nie, C. Chen, X. Jiang and J.-M. Liu, *Appl. Phys. Lett.*, 2021, **119**, 012405.
- 21 L. S. Xie, S. Husremovic, O. Gonzalez, I. M. Craig and D. K. Bediako, *J. Am. Chem. Soc.*, 2022, **144**, 9525–9542.
- 22 K. Niu, G. Qiu, C. Wang, D. Li, Y. Niu, S. Li, L. Kang, Y. Cai, M. Han and J. Lin, *Adv. Funct. Mater.*, 2023, **33**, 2208528.
- 23 M. Kan, B. Wang, Y. H. Lee and Q. Sun, *Nano Res.*, 2015, **8**, 1348–1356.
- 24 X. Liu, A. P. Pyatakov and W. Ren, *Phys. Rev. Lett.*, 2020, **125**, 247601.
- 25 T. Jungwirth, X. Marti, P. Wadley and J. Wunderlich, *Nat. Nanotechnol.*, 2016, **11**, 231–241.
- 26 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 27 M. Ernzerhof and G. E. Scuseria, *J. Chem. Phys.*, 1999, **110**, 5029–5036.
- 28 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 29 G. J. Martyna, M. L. Klein and M. Tuckerman, *J. Chem. Phys.*, 1992, **97**, 2635–2643.
- 30 S. L. Dudarev, G. A. Botton, S. Y. Savrasov, C. J. Humphreys and A. P. Sutton, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **57**, 1505.
- 31 S. Sarkar and P. Kratzer, *Phys. Rev. B*, 2021, **103**, 224421.

- 32 T. O. Wehling, A. I. Lichtenstein and M. I. Katsnelson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 235110.
- 33 X. Zhang, Z. Bao, X. Ye, W. Xu, Q. Wang and Y. Liu, *Nanoscale*, 2017, **9**, 11231.
- 34 J. Taylor, H. Guo and J. Wang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2001, **63**, 245407.
- 35 P. Panigrahi, P. K. Panda, Y. Pal, H. Bae, H. Lee, R. Ahuja and T. Hussain, *ACS Appl. Nano Mater.*, 2022, **5**, 2984–2993.
- 36 J. Prasongkit, E. de Freitas Martins, F. A. L. de Souza, W. L. Scopel, R. G. Amorim, V. Amornkitbamrung, A. R. Rocha and R. H. Scheicher, *J. Phys. Chem. C*, 2018, **122**, 7094–7099.
- 37 S. Husremović, C. K. Groschner, K. Inzani, I. M. Craig, K. C. Bustillo, P. Ercius, N. P. Kazmierczak, J. Syndikus, M. V. Winkle, S. Aloni, T. Taniguchi, K. Watanabe, S. M. Griffin and D. K. Bediako, *J. Am. Chem. Soc.*, 2022, **144**, 12167–12176.
- 38 K. J. Koski, J. J. Cha, B. W. Reed, C. D. Wessells, D. Kong and Y. Cui, *J. Am. Chem. Soc.*, 2012, **134**, 7584–7587.
- 39 X. Zhao, P. Song, C. Wang, A. C. Riis-Jensen, W. Fu, Y. Deng, D. Wan, L. Kang, S. Ning, J. Dan, T. Venkatesan, Z. Liu, W. Zhou, K. S. Thygesen, X. Luo, S. J. Pennycook and K. P. Loh, *Nature*, 2020, **581**, 171–177.
- 40 H. Bai, Q. Wu, H. Ai, D. Liu, J. Feng, L. K. Ang, Y. Lu, M. Yang and H. Pan, *Adv. Electron. Mater.*, 2022, **8**, 2200209.