



Cite this: *Chem. Commun.*, 2025, 61, 17302

## Semiconductor photocatalysts for hydrogen evolution: critical role of cocatalysts in enhancing performance

Bhagyashree Priyadarshini Mishra,\* Jasbir Dahiya and Venkata Krishnan \*

The conversion of abundant solar energy into green H<sub>2</sub> fuel from photocatalytic water splitting has been extensively studied to address the growing energy demand. In this context, various semiconductors have been employed as photocatalysts; however, insufficient active sites for the redox reactions limit H<sub>2</sub> evolution posing a significant challenge in semiconductor photocatalysis. Often, the use of a tiny amount of an additional constituent, known as a 'cocatalyst', can solve the issue by synergistically enhancing the performance of the semiconductor. Over the past few decades, noble metals and their derivatives have attracted huge research interest as cocatalysts; nevertheless, their limited availability and high cost significantly hinder large-scale applications. Hence, tremendous research attention has been devoted recently to design and develop inexpensive, earth-abundant, and highly stable materials which can be used as H<sub>2</sub> evolution cocatalysts. This review article gives an outline of the recent advances in the high performance materials that have been used as efficient H<sub>2</sub> evolution cocatalysts alongside photoactive semiconductors. Specifically, the H<sub>2</sub> evolution cocatalysts summarized in this article are classified under metal nanoparticles, single atoms, bimetallic alloys, metal oxides, transition metal dichalcogenides, metal phosphides, metal carbides, metal borides and carbon based cocatalysts. Finally, the article discusses the pivotal challenges that restrict future advancements and provide potential research directions in the field of cocatalyst-driven photocatalytic H<sub>2</sub> evolution using active semiconductors.

Received 4th August 2025,  
Accepted 1st October 2025

DOI: 10.1039/d5cc04459f

[rsc.li/chemcomm](http://rsc.li/chemcomm)

*School of Chemical Sciences and Advanced Materials Research Center, Indian Institute of Technology Mandi, Kamand, Mandi, 175075, Himachal Pradesh, India.*  
*E-mail: bhagyashree94dkl@gmail.com, vkn@iitmandi.ac.in*



**Bhagyashree Priyadarshini Mishra**

Her current research focuses on the development of effective photocatalysts for hydrogen evolution, nitrogen fixation and carbon dioxide reduction.

*Dr Bhagyashree Priyadarshini Mishra received her MSc degree from Ravenshaw University, Odisha, India. She completed her PhD from Siksha 'O' Anusandhan University, Odisha, where she worked in the field of photocatalytic hydrogen evolution and H<sub>2</sub>O<sub>2</sub> production. She joined the research group of Prof. Venkata Krishnan in 2024 as a postdoctoral researcher under the National Postdoctoral Fellowship (NPDF) scheme of Government of India. Her*



**Jasbir Dahiya**

include heterogeneous catalysis, photocatalysis, and defect engineering strategies for sustainable energy conversion.

*Jasbir Dahiya received his BSc degree in Chemistry from M.D.U. University, Haryana, in 2022 and completed his MSc degree in Chemistry from the Indian Institute of Technology Mandi, India, in 2025. During his master's degree, he worked on the design of defect-rich transition metal oxide catalysts for photocatalytic hydrogen evolution and nitrogen fixation under the supervision of Prof. Venkata Krishnan. His research interests*



# 1. Introduction

Rapid industrialization has increased the demand for fossil fuels like coal, oil, and natural gas, which has put a heavy burden on the environment and led to a sharp increase in greenhouse gas emissions.<sup>1–3</sup> A key tactic for solving the energy problem and environmental issues is switching to clean and renewable energy sources.<sup>3–5</sup> Hydrogen (H<sub>2</sub>) is increasingly recognized as a clean and sustainable energy source, making it a strong contender to replace carbon-based fuels.<sup>6,7</sup> There are several ways to produce H<sub>2</sub>, including water electrolysis, reforming natural gas, converting biomass, reforming methanol, and breaking down ammonia.<sup>8–12</sup> Each of these methods has its own benefits and challenges, adding to the growing appeal of H<sub>2</sub> as an environmentally friendly energy option. In this regard, solar energy is noteworthy because of its enormous availability and potential for a variety of uses. Overall splitting of water utilizing solar energy results in the evolution of both H<sub>2</sub> and O<sub>2</sub> in a stoichiometric ratio of 2 : 1 and is considered an attractive aspect that contributes to energy revolution.<sup>13–16</sup> However, focusing on selective H<sub>2</sub> production from water splitting is one potential technique that not only provides a way to convert sunlight into clean chemical energy but also offers a practical solution to the current energy and environmental issues.<sup>17–19</sup>

In this regard, after the revolutionary research by Fujishima and Honda in 1972 that water could be split into O<sub>2</sub> and H<sub>2</sub> using an electrochemical cell with a TiO<sub>2</sub> electrode, illuminated by ultraviolet light, and coupled to a platinum electrode without the need for an external voltage, numerous UV active semiconducting materials have been utilized for photocatalysis.<sup>20,21</sup> The photocatalytic process typically entails the absorption of light followed by the migration of photoexcited electron-hole pairs to surface active sites of the semiconductor, where they initiate reduction and/or oxidation reactions. The equilibrium of the thermodynamics and kinetics of the charge generation and

separation process determines the overall efficiency of photocatalytic water splitting.<sup>22</sup> Moreover, the efficiency of these materials is constrained by their narrow light absorption, mostly limited to the ultraviolet region, which accounts for just 5% of the solar spectrum, and their high rate of charge carrier recombination, creating a significant barrier to practical implementation. Thus, it is essential to enhance the absorption capabilities of the wide-bandgap materials by boosting light absorption, suppressing the recombination of photoinduced electron-hole pairs, and expanding their sensitive wavelength range.

Therefore, various strategies have been employed to obtain highly efficient semiconductor photocatalysts with increased potential for selective H<sub>2</sub> evolution from aqueous solution, like bandgap engineering (by doping metals/non-metals), modulating size and morphology, coupling with various narrow-bandgap semiconductors, and incorporating cocatalysts that exhibit localized surface plasmon resonance (SPR) effects.<sup>23–27</sup> Among these techniques, the combination of semiconductors with efficient cocatalysts has attracted a lot of interest nowadays. So far, a variety of cocatalysts have been used in photocatalytic H<sub>2</sub> evolution reactions, with the majority of attention focused on noble metal-based cocatalysts. Elements such as Pt, Pd, Au, Ag, and Ru have been thoroughly studied and found to significantly improve the photocatalytic H<sub>2</sub> evolution performance of the semiconductor.<sup>27</sup> Furthermore, cocatalysts derived from noble metals, such as PtS and Rh<sub>x</sub>P, have garnered a lot of interest because of their higher catalytic activity compared to their elemental counterparts.<sup>28,29</sup> However, the scarcity and high cost of these precious metals prevent their widespread application. Because of this, increasing attention is being directed toward affordable cocatalysts made of earth-abundant elements that can act as electron sinks to enable charge migration and encourage hydrogen evolution. These cocatalysts include metal nanoparticles, single atoms, bimetallic alloy cocatalysts, metal oxide cocatalysts, transition metal dichalcogenide cocatalysts, metal phosphide cocatalysts, metal carbide cocatalysts, carbon based cocatalysts, and metal boride cocatalysts obtained from inexpensive and earth-abundant elements that can show their excellence in photocatalytic H<sub>2</sub> evolution.

With the recent surge in advancements related to the development of cost-effective, stable, and efficient cocatalysts, it is an opportune moment to systematically evaluate and summarize photocatalytic systems where semiconductors are integrated with cocatalysts. Some impressive reports have summarized the synthesis of cocatalysts with their loading process over the support semiconductor, along with their photocatalytic applications in various fields like carbon dioxide reduction, nitrogen fixation, and pollutant degradation. Similarly, different cocatalyst loading over a particular support material has been summarized recently. Moreover, reviews summarizing the localized surface plasmon effect based cocatalysts have also been reported. However, a comprehensive assessment concerning the critical role of various cocatalysts over well-known support materials in photocatalytic H<sub>2</sub> evolution is scanty. In this

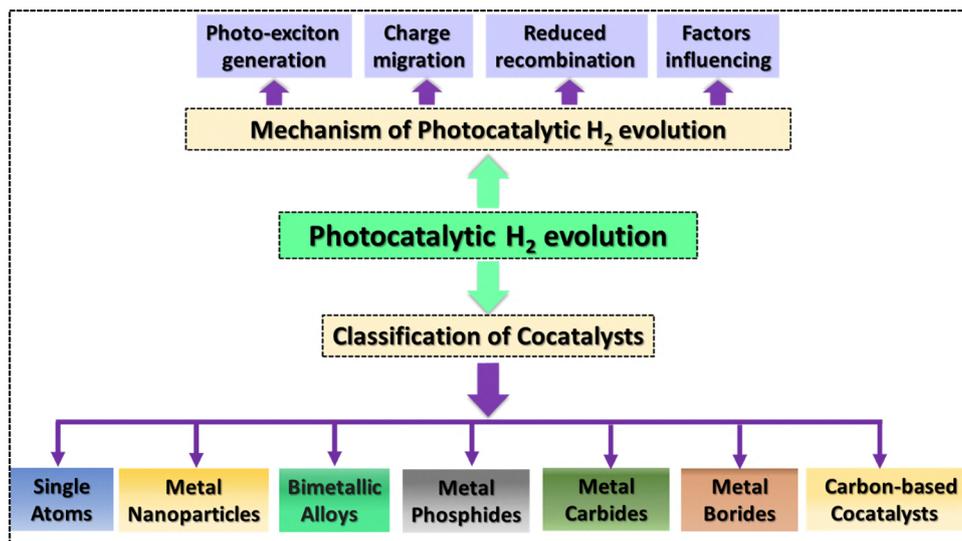


**Venkata Krishnan**

*Prof. Venkata Krishnan obtained his PhD from the University of Stuttgart, Germany, in July 2006. Later, he worked as a postdoctoral researcher at the University of Pennsylvania, USA, from 2006 to 2010 and then as a research associate at the National Institute for Materials Science, Japan, from 2010 to 2012. He joined IIT Mandi in 2012 and is currently working as a Full Professor in the School of Chemical Sciences. His research*

*interest is in the broad field of green chemistry and heterogeneous catalysis for energy and environmental applications.*





Scheme 1 Schematic illustration of various aspects discussed in this article.

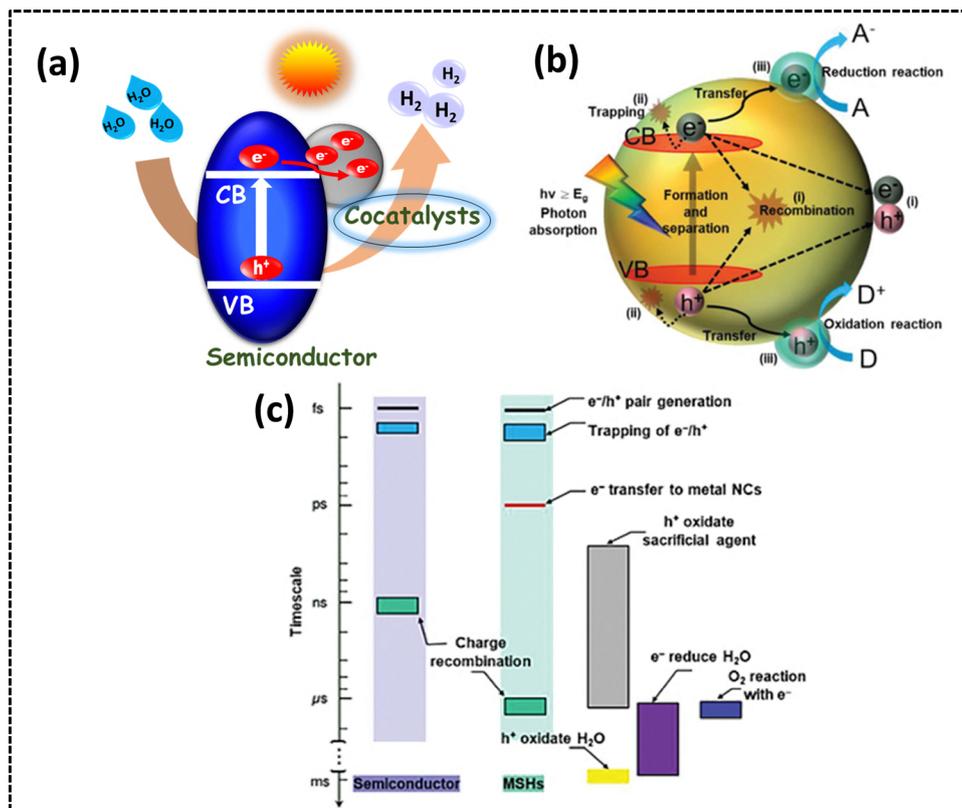
article, the contribution of cocatalysts in enhancing the photocatalytic H<sub>2</sub> evolution from water splitting is discussed in detail. Furthermore, the cocatalysts have been classified as metal nanoparticles, single atoms, bimetallic alloys, metal oxides, transition metal dichalcogenides, metal phosphides, metal carbides, metal borides and carbon based cocatalysts and their modification with various widely used semiconductors is categorically discussed. Advantages as well as disadvantages of the cocatalysts have been discussed briefly. Finally, an assessment of the present state and future prospects of the field is given to stimulate creative ideas for developing cocatalyst-modified heterojunction systems in catalytic applications within this rapidly expanding area. This study thus concentrates on new developments in the use of cocatalysts in photocatalytic H<sub>2</sub> evolution reactions. The various cocatalysts utilized for photocatalytic H<sub>2</sub> evolution and covered in this article are shown in Scheme 1.

## 2. Mechanistic insight into improved photocatalytic activity by cocatalysts

The basic principle of photocatalysis involves absorption of photons by the semiconducting material with an appropriate bandgap, resulting in the generation of photogenerated charge carriers (electrons and holes). These photoexcited charge carriers then migrate to the surface to participate in various redox reactions. The sequential steps after photo-exciton generation are represented in Scheme 2a and b. The photocatalytic hydrogen evolution process can be divided into three primary stages: electron generation, electron transport, and electron utilization. The initial stage is electron generation in which the photocatalyst, typically in powder form, is dispersed in an aqueous medium. Upon light irradiation, electrons in the valence band (VB) are excited to the conduction band (CB), leaving behind holes in the VB. For efficient photocatalysis, the

material should exhibit strong light absorption over a broad wavelength range and a high quantum yield. Additionally, the CB edge must be more negative than the standard reduction potential of H<sup>+</sup>/H<sub>2</sub> to ensure that the excited electrons possess sufficient reducing power. The subsequent step is electron transport. After excitation, the charge carriers migrate to the surface of the photocatalyst. However, some electrons and holes recombine within the bulk or on the surface, leading to energy loss as heat or light. This recombination significantly diminishes overall efficiency. To counteract this, effective spatial charge separation is essential. Achieving this separation under a suitable driving force remains one of the main challenges in photocatalytic systems. Scheme 2c represents the respective timescales for the photoinduced charge carriers. The final step is electron utilization. Photocatalytic water splitting using a semiconducting material denotes the stoichiometric splitting of water into H<sub>2</sub> and O<sub>2</sub> in the absence of sacrificial agents. To promote H<sub>2</sub> evolution in aqueous medium, a half reaction of water splitting, electron donors or hole scavengers have been used to irreversibly oxidize the reducing agent instead of water. Basically, surface holes are effectively scavenged by electron donors to prevent recombination. The most commonly used hole scavengers include alcohols (*e.g.*, methanol, ethanol, ethylene glycol, glycerol), organic acids (*e.g.*, ascorbic, lactic, acetic), EDTA salts, sulfide species (Na<sub>2</sub>S–Na<sub>2</sub>SO<sub>3</sub>), and various reducing agents like glucose, triethanolamine, and triethylamine. Subsequently, electrons will be available over the surface of the photocatalyst. Hence, the retained electrons then drive the hydrogen evolution reaction, reducing protons to molecular hydrogen. For this step to proceed efficiently, the electron potential must be sufficiently negative, and the system must overcome kinetic overpotentials. It has also been proven that the Gibbs energy involved in hydrogen evolution accompanied with the oxidative decomposition of the sacrificial reagent is typically smaller than that of overall water splitting. This means that the energy stored in





**Scheme 2** (a) Illustration of the photocatalytic hydrogen evolution mechanism involving a cocatalyst-loaded semiconductor, (b) depiction of the sequential processes like light absorption, charge carrier separation, migration, and recombination that occur within a cocatalyst-modified semiconductor during photocatalytic hydrogen production, (c) respective timescales in unmodified semiconductors and in metal–semiconductor heterostructures. Reproduced with permission.<sup>40</sup> Copyright 2021, Wiley.

'hydrogen' obtained using a sacrificial reagent is smaller compared to the case of hydrogen obtained from overall water splitting. Therefore, the ultimate goal is  $H_2$  production using water as a hydrogen and electron source. Moreover, the  $H_2$  evolution rate is influenced by several factors, including the number of available active sites, the specific reaction pathway, and mass transport limitations at the solid–liquid interface.<sup>4,30</sup> Hence, to enhance  $H_2$  evolution from water splitting without using sacrificial agents, it is essential to utilize the photoinduced surface electrons and this is being explored by the scientific community.<sup>31</sup>

As discussed before, photocatalytic water splitting causes both reduction of water to produce  $H_2$  and oxidation of water to produce  $O_2$ , and back reaction between  $H_2$  and  $O_2$  to form water is possible as it is an uphill reaction. This can also affect the overall efficiency of photocatalytic  $H_2$  evolution. Hence, in order to increase the  $H_2$  evolution quantum efficiency, photocatalysts with surface properties that suppress back reactions are required. In this regard, cocatalysts are being loaded to accelerate the transfer of photogenerated electrons from the conduction band of the semiconductor to the surface of active cocatalysts, thereby achieving effective electron–hole separation. In addition, cocatalysts can reduce the activation energy or overpotential required for hydrogen evolution reactions occurring on semiconductor surfaces.<sup>1,32</sup> Another important role of

cocatalysts is mitigating photo-corrosion and enhancing the stability of semiconductor photocatalysts. This cocatalyst-induced mechanism for enhancing charge transfer can be categorized into two distinct components:

Firstly, the integration of noble metals or metallic cocatalysts with semiconductor photocatalysts can induce the formation of Schottky barriers at the interface, which generate an internal electric field that facilitates efficient charge separation. This process is driven by the variance in Fermi levels between the metal cocatalysts and the semiconductor, resulting in charge redistribution until Fermi level alignment is achieved. The ensuing band bending and Schottky barrier formation effectively suppress the recombination of photogenerated charge carriers by hindering the backflow of electrons. This promotes the participation of a greater number of electrons in reduction reactions such as hydrogen evolution. Furthermore, certain noble metals, including Au and Ag, exhibit localized surface plasmon resonance (LSPR), which enhances light absorption and contributes to increased charge carrier utilization. In addition, cocatalysts composed of precious metals (e.g.,  $PtS_2$ , RuP), transition-metal phosphides, sulfides, and carbides are also capable of forming Schottky junctions, thereby further improving interfacial charge separation and photocatalytic performance. Secondly, conductive non-metallic carbon-based materials such as graphene and carbon nanotubes have the



ability to form Schottky junctions which can also improve the charge separation and enhance the photocatalytic hydrogen evolution.<sup>33,34</sup>

However, the efficiency of cocatalysts in photocatalytic hydrogen evolution reactions is influenced by several factors, including the method of deposition, loading amount, particle size, morphology, dispersion uniformity, and chemical composition.<sup>35</sup> Consequently, there is typically an optimal set of parameters for cocatalyst application that leads to the highest photocatalytic performance. For instance, in the case of cocatalyst loading amount, insufficient deposition can lead to suboptimal separation of photogenerated charge carriers, thereby limiting H<sub>2</sub> evolution efficiency. Conversely, excessive loading may hinder photocatalytic activity by blocking active surface sites, reducing light absorption, and causing non-uniform particle dispersion. Morphology of the cocatalysts employed also affects the activity of the overall semiconductor–cocatalyst system. In this regard, Park *et al.* studied how the morphology of Pt (round, cube, and rough shape) over CdSe nanorods affects photocatalytic H<sub>2</sub> evolution. They found that the rough tip Pt cocatalyst showed the highest photocatalytic H<sub>2</sub> evolution followed by round and cubic tips.<sup>36</sup> In addition, the facets over which a cocatalyst is loaded are also crucial in determining the efficiency of photocatalytic processes. Different crystal facets of a semiconductor often preferentially accumulate either photogenerated electrons or holes due to intrinsic anisotropic charge transport. For instance, photogenerated electrons and holes preferentially migrate to distinct facets of BiVO<sub>4</sub> – electrons accumulate on the {010} facets, while holes accumulate on the {110} facets. This spatial separation enables reduction reactions to occur on {010} facets and oxidation reactions to occur on {110} facets. Hence, when the cocatalysts are selectively deposited on their corresponding active facets, with reduction cocatalysts on {010} and oxidation cocatalysts on {110}, the charge carriers are effectively trapped and utilized, minimizing recombination losses and significantly enhancing photocatalytic and photoelectrochemical activity. In contrast, random or misaligned deposition of cocatalysts leads to inefficient charge separation, lower activity, and even suppressed performance.<sup>37</sup> In another study, it was shown that by selectively loading Rh/Cr<sub>2</sub>O<sub>3</sub> reduction cocatalysts on electron-rich facets and CoOOH oxidation cocatalysts on hole-rich facets, the hydrogen evolution reaction and oxygen evolution reaction can be enhanced separately.<sup>38</sup> Moreover, it has been reported that local heat generated by light absorption by cocatalyst nanoparticles may positively affect the photocatalytic activity. When plasmonic or light-absorbing cocatalyst nanoparticles (such as Au, Ag, or certain transition metal oxides) interact with incident photons, they not only facilitate charge transfer but also convert part of the absorbed light into localized thermal energy. This nanoscale heating can enhance surface reaction kinetics by lowering the activation energy barriers of redox processes and improving the mobility of reactants and intermediates near the active sites. Furthermore, the photothermal effect can promote desorption of reaction products from the catalyst surface, preventing site blockage and

sustaining activity. Therefore, cocatalysts with dual photonic and photothermal properties represent a promising strategy to synergistically boost photocatalytic efficiency.<sup>39</sup>

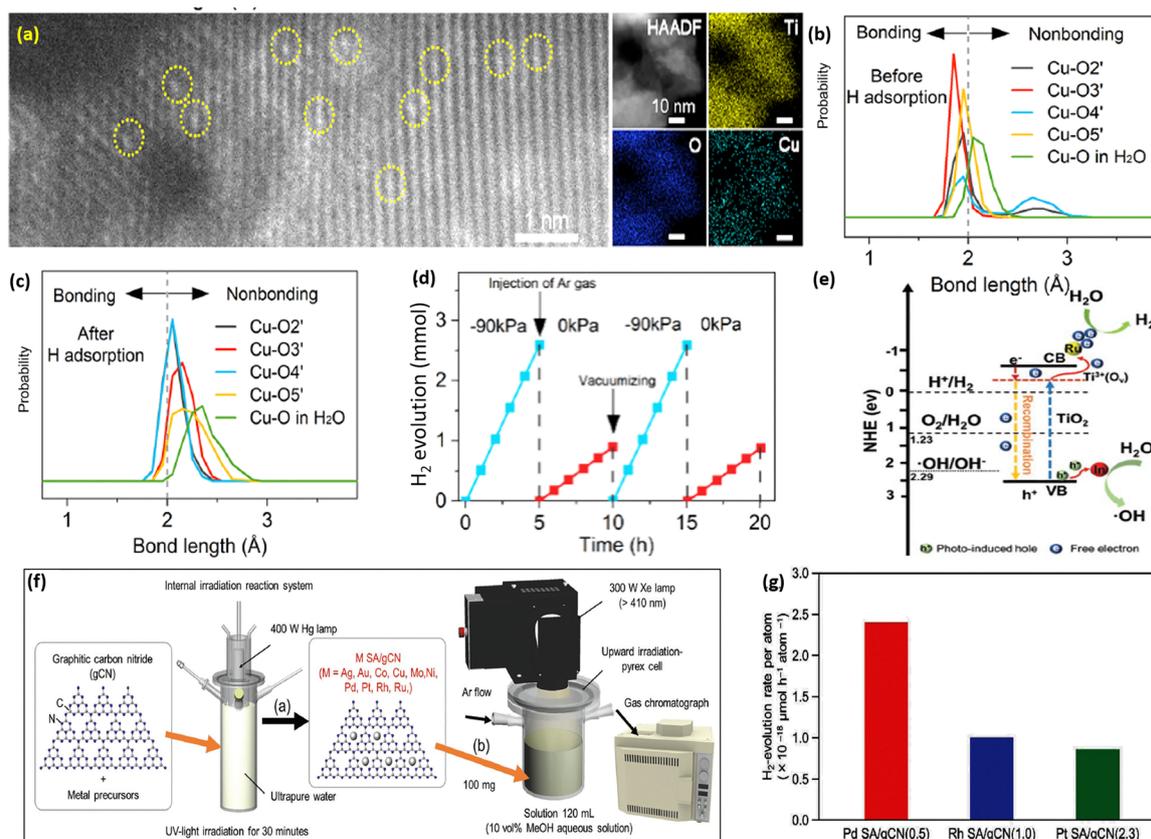
### 3. Classification of co-catalysts for enhanced photocatalytic H<sub>2</sub> evolution

#### 3.1. Single-atoms as co-catalysts

Recently, single-atom (SA) metals with an even smaller size than metal NPs have emerged as highly promising cocatalysts in photocatalysis due to their unsaturated coordination environments and exceptional catalytic activity, offering significant potential for applications such as hydrogen evolution. Notably, metal SA catalysts can exhibit high catalytic performance while significantly reducing material usage and overall cost. Unlike, metal nanoparticles where the atoms inside the nanoparticles cannot contribute to the reaction as the catalytic reactions occur on the surface, SAs, which lack inactive internal atoms, can utilize every metal atom as an active site, significantly enhancing the catalytic performance on a per-metal-atom basis. Hence, SACs decorated over semiconductors have been constructed for enhancement of photocatalytic H<sub>2</sub> evolution.<sup>41,42</sup> In this context, Fu *et al.* synthesized Cu SACs supported on the (101) surface of TiO<sub>2</sub>, in which the incorporation of Cu atoms in place of Ti vacancies was confirmed from the HAADF-STEM imaging, as shown in Fig. 1a.<sup>43</sup> They explored that the Cu atoms in Cu-SAC/TiO<sub>2</sub> change their coordination during photocatalytic hydrogen evolution. In addition, it is reported that the local Cu–O structure plays a vital role in boosting catalytic activity as shown in Fig. 1b and c. Simulations reveal that the Cu d<sub>2</sub> orbital efficiently captures photogenerated electrons and transfers them to water molecules, driving the H<sub>2</sub> evolution reaction. Briefly, the generated H atom during the Volmer process adsorbed to a neighboring bridging O atom of the Cu SAC, which led to changes in the valence state as well as the color of Cu. It is concluded that the adsorption of the H atom is not responsible for the activation of Cu-SAC/TiO<sub>2</sub>; instead, it creates an inert state by adding an extra electron into the Cu d<sub>2</sub> orbital. This single atom decorated metal oxide exhibited higher stability towards the photocatalytic H<sub>2</sub> evolution even when subjected to varying pressure and irradiation over a 20-h period, as shown in Fig. 1d.

Similarly, Peng *et al.* introduced an advanced photocatalyst, Ru/In dual-single atoms doped on TiO<sub>2</sub> nanosheets (Ru–In SA/TiO<sub>2</sub>), designed to address the challenges of efficient hydrogen production through photocatalytic water splitting without sacrificial agents. This novel catalyst outperformed its counterparts such as Ru-doped, In-doped, and pristine TiO<sub>2</sub> by 6, 18, and 53 times, respectively, with a remarkable hydrogen evolution rate of 174.1 μmol h<sup>−1</sup>. The dual-single atom doping approach increased the separation of photogenerated electrons and holes enhancing charge carrier mobility and reducing recombination by boosting Ti<sup>3+</sup> sites and oxygen vacancies. Moreover, Ru facilitated electron transfer and In enhanced hole transfer, which inhibited electron–hole recombination and





**Fig. 1** (a) High-resolution STEM and STEM-EDS mapping of Cu-SAC/TiO<sub>2</sub>. (b) and (c) Cu–O bond length distributions before and after hydrogen adsorption at 300 K. The observed elongation of Cu–O bond lengths after hydrogen adsorption indicates that the electron donated by the hydrogen atom enters Cu atom's d orbital, resulting in a repulsive interaction between the Cu atom and the H<sub>2</sub>O molecule. (d) Photocatalytic H<sub>2</sub> evolution of Cu-SAC/TiO<sub>2</sub> under varying dynamic pressure in an Ar-protected environment, with 0 kPa being the standard atmospheric pressure. Reproduced with permission.<sup>43</sup> Copyright 2023, American Chemical Society. (e) The tentative mechanism for photocatalytic H<sub>2</sub> production over Ru–In SA/TiO<sub>2</sub>. Reproduced with permission.<sup>44</sup> Copyright 2022, Wiley. (f) (a) Synthesis of M SA-loaded g-CN and (b) photocatalytic H<sub>2</sub> evolution for M SA/g-CN photocatalysts. (g) H<sub>2</sub> evolution rate per metal atom for M SA/g-CN photocatalysts. Reproduced with permission.<sup>45</sup> Copyright 2023, Wiley.

accounted for this efficiency (Fig. 1e). The catalyst provides an affordable and environmentally friendly way to produce hydrogen, which aligns with the goals of renewable energy. It also functions effectively without a sacrificial agent and shows remarkable stability.<sup>44</sup>

Furthermore, Akinaga and co-workers explored the importance of single-atom cocatalysts on graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) for enhancing the photocatalytic H<sub>2</sub> evolution reaction.<sup>45</sup> They examined ten different transition metal elements (Ag, Au, Co, Cu, Mo, Ni, Pd, Pt, Rh, and Ru) supported on g-C<sub>3</sub>N<sub>4</sub>, by a photo-deposition method to achieve high weight loading ( $\approx 3$  wt%) while preventing aggregation as shown in Fig. 1f. Experimental results demonstrated that Pd, Pt, and Rh single-atom catalysts supported on g-C<sub>3</sub>N<sub>4</sub> exhibited superior H<sub>2</sub> evolution activity, with Pd SA catalysts achieving the highest H<sub>2</sub> evolution rate per active site owing to their favorable electronic configuration, which enhances proton adsorption and facilitates efficient charge transfer. Transient absorption spectroscopy, electrochemical studies, and X-ray absorption fine structure all verified that SA cocatalysts effectively promote charge separation and considerably lower the H<sub>2</sub> evolution

overpotential. It is noteworthy that g-C<sub>3</sub>N<sub>4</sub> loaded with Pd SAs demonstrated an 8.6-fold increase in H<sub>2</sub> evolution efficiency (Fig. 1g) per active site in comparison to Pd nanoparticles, demonstrating the benefits of single-atom dispersion in maximizing photocatalytic hydrogen generation. In a subsequent study, Ren and co-workers studied the effect of particle size of co-catalysts on photocatalytic H<sub>2</sub> evolution.<sup>46</sup> According to the study, the H<sub>2</sub> evolution activity of the covalent organic layers supported on SiO<sub>2</sub> nanoparticles was greatly increased when Pd nanoparticles were downsized to the atomic or cluster scale (Fig. 2a). An enhanced hydrogen evolution rate with an apparent quantum efficiency of 7.3% was observed with the Pd SAs/Cs cocatalyst. The enhancement in photocatalytic activity was attributed to the enhanced charge separation efficiency of COF layers rather than proton reduction.

A unique Z-scheme single atom photosystem has been developed by Liang *et al.* which appears as a triangular prism shaped metal-organic cage (MOC-Q3) embedded with 3 catalytic Pd<sup>2+</sup> sites and two photosensitive ligands, and the MOC-Q3 was loaded over a highly crystalline  $\beta$ -ketoenamine-linked COF (EA-COF) as shown in Fig. 2b.<sup>47</sup> The optimized catalyst,



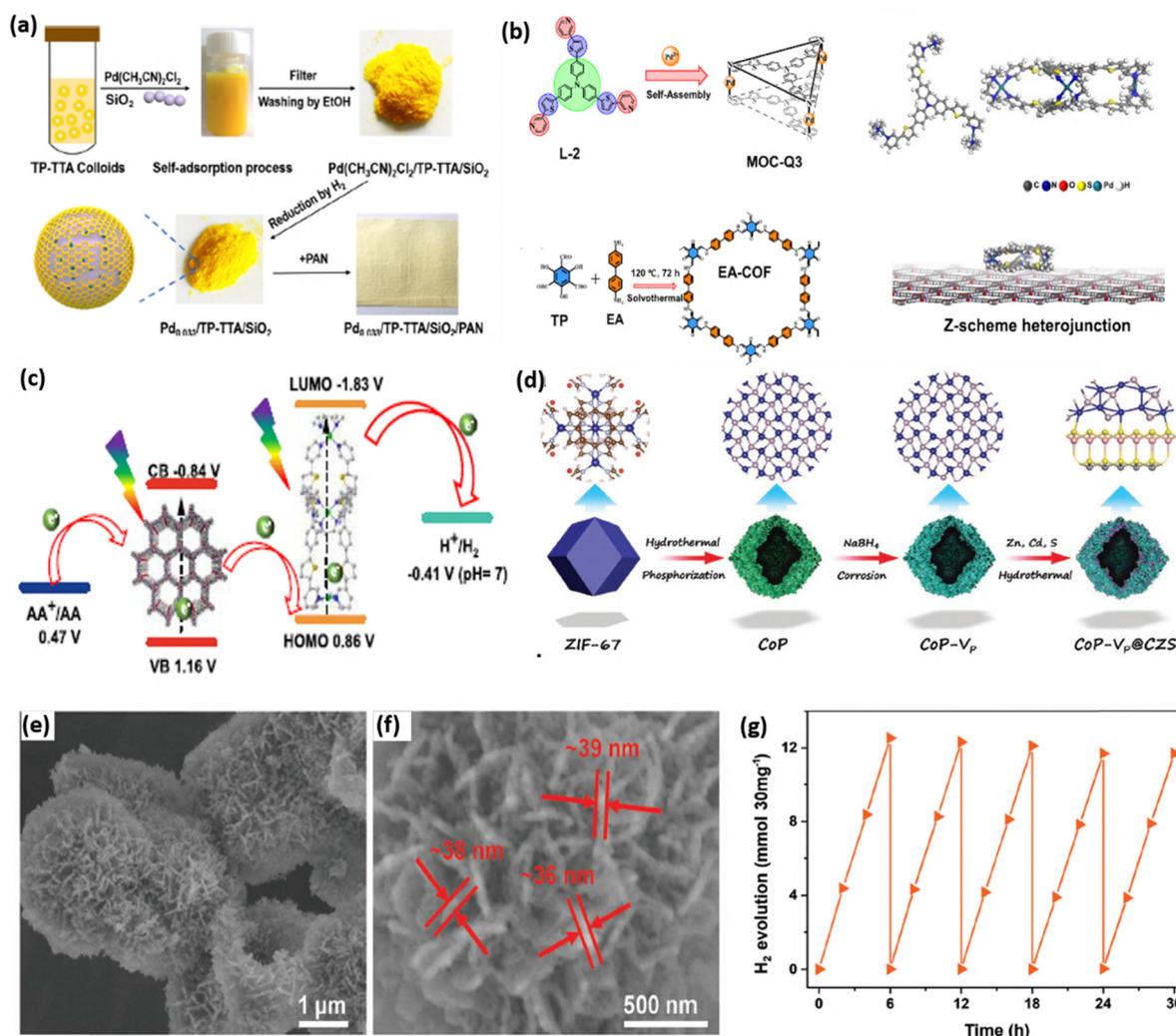


Fig. 2 (a) Schematic illustration of the synthesis of Pd SAs/Cs on TP-TTA COF layers and the photoactive Pd<sub>0.033</sub>/TP TTA/SiO<sub>2</sub>/PAN film. Reproduced with permission.<sup>46</sup> Copyright 2022, American Chemical Society. (b) Assembly of MOC-Q3 using ligand L-2 and Pd<sup>2+</sup>, crystal structure of MOC-Q3, EA-COF preparation using TP and EA, and structure of MOC-Q3/EA-COF. (c) Working mechanism of the MOC-Q3/EA-COF system. Reproduced with permission.<sup>47</sup> Copyright 2024, American Chemical Society. (d) Schematic representation of the synthesis of the CoP-Vp@CZS photocatalyst. (e) FESEM images. (f) High magnification image of CoP-Vp@CZS. (g) Recyclability study of CoP-Vp@CZS in successive H<sub>2</sub> evolution cycles. Reproduced with permission.<sup>48</sup> Copyright 2023, Wiley.

MOC-Q3/EA-COF, was observed to achieve a higher H<sub>2</sub> evolution with a TON<sub>Pd</sub> of 118 521 when ascorbic acid was used as a sacrificial agent. Furthermore, the H<sub>2</sub> evolution rate for the Z-scheme system increased about 1.8 times under piezophotocatalytic conditions. The enhanced photocatalytic performance was due to the broad light absorption, effective charge separation, and uniform distribution of the Pd SA active sites. The MOC-Q3/EA-COF Z-scheme photocatalyst also showed better photocurrent and less charge transfer resistance compared to the other counterparts. The photocatalytic mechanism was revealed by calculating the band edge potentials combining the outputs from Mott-Schottky measurements, valence band X-ray photoelectron spectroscopy, and UV-vis DRS as shown in Fig. 2c.

Zhang and co-workers reported a novel strategy to enhance photocatalytic hydrogen production by engineering single-atom

phosphorus vacancies into the CoP cocatalyst (CoP-V<sub>p</sub>) and the cocatalyst was coupled with Cd<sub>0.5</sub>Zn<sub>0.5</sub>S to form the CoP-V<sub>p</sub>@Cd<sub>0.5</sub>Zn<sub>0.5</sub>S (CoP-Vp@CZS) heterojunction.<sup>48</sup> The phosphorus defect enriched CoP hollow microspheres were synthesized from phosphorization of ZIF-67 followed by the liquid phase corrosion process as shown in Fig. 2d. The CoP obtained from phosphorization of the intermediate Co-OH had a hollow structure composed of a uniform assembly of vertically aligned nanoflakes (Fig. 2e and f). However, anchoring of CZS increased the thickness of the nanoflakes to 38 nm. In photocatalytic H<sub>2</sub> evolution, the CoP-Vp@CZS catalyst achieved a remarkable hydrogen evolution rate of 2.052 mol h<sup>-1</sup> per 30 mg of catalyst, which was 1.41 and 14.66 times greater than that of the pure CoP@CZS and ZCS samples, respectively, and a little decline in activity was noticed after 40 h of uninterrupted stability tests as shown in Fig. 2g. The enhanced H<sub>2</sub> evolution



was supported by the results obtained from photocurrent and time-resolved photoluminescence analysis. Hence, the incorporation of single atom CoP-Vp on to the CZS semiconductor successfully enhanced the photocatalytic H<sub>2</sub> evolution efficiency. Similarly, Zhao *et al.* anchored cobalt single atoms over nitrogen-doped graphene (NG) to create an effective cocatalyst (Co-NG).<sup>49</sup> They found that the photocatalytic H<sub>2</sub> evolution increased to 1382 μmol h<sup>-1</sup> when Co-NG was incorporated with CdS which is 3.42 times higher than that achieved with the NG-loaded CdS photocatalyst. This study also proved that SACs improve photocatalytic H<sub>2</sub> evolution reaction. In another study, Zeng and coworkers fabricated Pt single atom incorporated carbon nitride with a significant content of 8.7 wt%. The photocatalyst exhibited a H<sub>2</sub> evolution rate of 22 650 μmol g<sup>-1</sup> h<sup>-1</sup> with 22.5% AQE (~420 nm). They found that the location of Pt SAs plays a crucial role in promoting the photocatalytic performance.<sup>50</sup> A series of Pd-loaded TiO<sub>2</sub> photocatalysts were synthesized by Lv *et al.*, suggesting that Pd size has a considerable impact on photocatalytic hydrogen evolution.<sup>51</sup> Amongst these, the maximum activity was shown by Pd<sub>0.75</sub>/TiO<sub>2</sub>, which was 15, 1.5, and 1.6 times greater than that of Pd<sub>f</sub>/TiO<sub>2</sub>, Pd<sub>0.5</sub>/TiO<sub>2</sub>, and Pd<sub>1.0</sub>/TiO<sub>2</sub>, respectively. DFT studies showed that the combination of sub-nano cluster and single-atom Pd (Pd<sub>SA</sub>) enhances the H<sup>+</sup> adsorption-desorption balance by optimising the d-band centre and bringing ΔG<sub>H<sup>+</sup></sub> to zero. In keeping with the noted greater charge separation, Pd<sub>SNC</sub> also encouraged electron transfer from Pd<sub>SA</sub>. The design of sophisticated photocatalysts was guided by this work, which provides important insights into photocatalytic H<sub>2</sub> evolution processes. Despite spectacular advantages of SACs in the field of photocatalytic H<sub>2</sub> evolution, they suffer from several shortcomings, which limits their potential as co-catalytic materials. Firstly, stability of SACs remains a critical issue as isolated metal atoms tend to migrate and aggregate under reaction conditions, leading to nanoparticle formation and loss of catalytic activity. Secondly, the limited density of active sites restricts overall reaction rates, especially for large-scale hydrogen production, since loading beyond a certain threshold often causes aggregation. Thirdly, the coordination environment of single atoms is highly sensitive to the support surface; hence any structural change during operation (*e.g.*, photo-corrosion or surface reconstruction) may drastically alter catalytic performance. Lastly, fabricating SACs with precise control over dispersion is a significant synthetic challenge.

### 3.2. Metal nanoparticles as co-catalysts

Metal nanoparticles significantly enhance photocatalytic efficiency due to their localized surface plasmon resonance (LSPR) effect, a current focus in energy research. Under resonant light excitation, free electrons in these nanocrystals oscillate collectively, generating intense electromagnetic fields and producing energetic “hot electrons”. The LSPR properties like strong light absorption and near-field enhancement are highly dependent on the nanoparticles’ shape, size, and composition. Pt nanoparticles, the highest performing H<sub>2</sub> evolution cocatalysts, exhibit work function above 5.5 eV. Other metal nanoparticles

show work function values in the range of 4.3–5.6 eV. Additionally, when plasmonic metals (*e.g.*, Au, Ag, Cu) contact n-type semiconductors, Fermi level alignment facilitates hot electron injection into the semiconductor’s conduction band, enabling kinetically demanding multielectron reactions by lowering activation barriers. Furthermore, metal nanoparticles are highly promising for H<sub>2</sub> evolution due to their large surface-to-volume ratio, abundance of active surface atoms, and distinct electronic properties compared to their bulk forms.<sup>52,53</sup>

In this regard, Zhang *et al.* effectively utilized the LSPR effect of Ag metal nanoparticles to utilize solar energy efficiently. They decorated the plasmonic Ag nanoparticles over a covalent organic framework by an *in situ* method (Fig. 3a). The photocatalyst is evidenced to achieve an outstanding H<sub>2</sub> evolution rate of 801.4 μmol g<sup>-1</sup> h<sup>-1</sup> with 3% loading of Ag NPs, which is nearly 4 times more than the rate of the pristine TpPa-1-COF. The red shift in the absorption edge compared to that of TpPa-1-COF as obtained from UV-Vis DRS indicates that the deposition of Ag NPs expands the visible-light responsive nature of the catalyst. The absorption peak at around 465 nm as depicted in Fig. 3b was identified due to the LSPR effect of Ag NPs. The hot electron generation was confirmed by finite-difference time-domain (FDTD) simulation study which showed that an increase in the local interfacial electric field significantly facilitates easy passage of hot electrons at the surface. The mechanistic study revealed that the hot electrons generated by the plasma action of Ag NPs flow back to the CB of TpPa-1-COF where the reduction process is carried out to form H<sub>2</sub> as the work function of n-type TpPa-1-COF is more than that of Ag NPs (Fig. 3c).<sup>54</sup>

In another study, Liu and coworkers prepared a series of Cu/In<sub>2</sub>O<sub>3</sub>/CdS photocatalysts to intrinsically regulate the photo-generated charge carrier transfer for elevated photocatalytic H<sub>2</sub> evolution (Fig. 3d). Metal nanoparticles like Au, Ag, *etc.* exhibit a LSPR effect that helps in broadening the light absorption range, refining charge separation, and facilitating the adsorption and dissociation of reactive molecules attributing to the generation of hot-electrons. Instead of using expensive NPs, herein, the cost-effective Cu NPs have been used to enhance the photocatalytic H<sub>2</sub> evolution by facilitating the reduction of H<sup>+</sup> to H<sub>2</sub>. The LMM spectra showed that the valence state of Cu in both binary and ternary composites is mainly Cu<sup>0</sup> instead of Cu<sup>+</sup>. The study reported a better photocatalytic H<sub>2</sub> evolution activity of CuNPs/In<sub>2</sub>O<sub>3</sub>/CdS (2840.25 μmol h<sup>-1</sup>) which exceeds the performance of pristine CdS based photocatalysts (Fig. 3e). The *in situ* XPS study verified the transfer of photoelectrons from CdS to Cu NPs as the binding energy of Cu 2p shifts negatively. The surface reaction thermodynamics confirmed the superior H<sup>+</sup> adsorption ability of Cu and the electron transfer from Cu to H<sup>+</sup>. The increased concentration of high-energy hot electrons, which enhances H<sub>2</sub> evolution, originated from two sources: one is the direct excitation due to the LSPR effect of Cu NPs and other is the transfer of photogenerated electrons from CdS to Cu NPs (Fig. 3f).<sup>55</sup>

Xu *et al.* synthesized a WS<sub>2</sub>@Cu hybrid by a two-step process in which Cu NPs are deposited over WS<sub>2</sub> by a photochemical



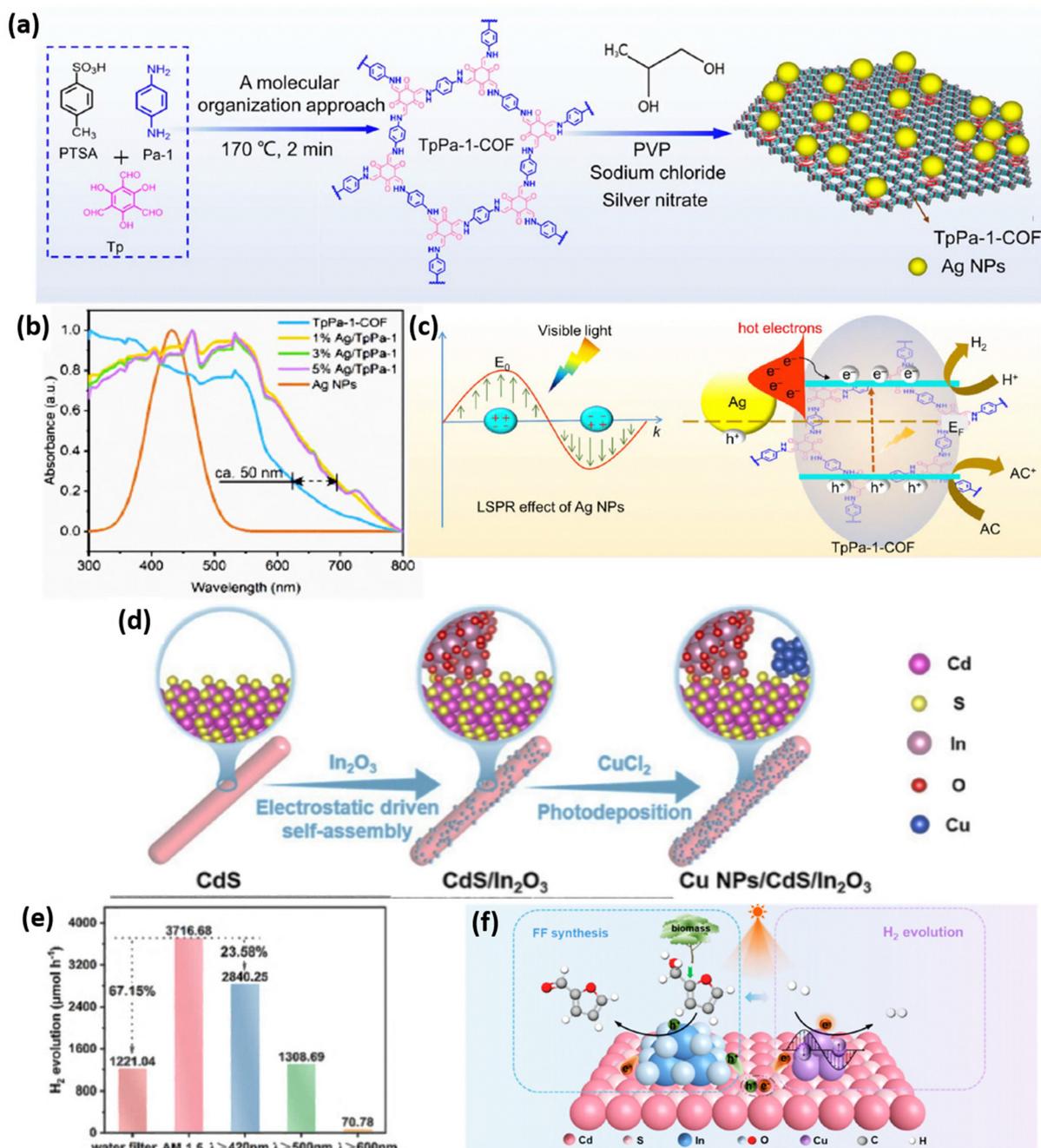
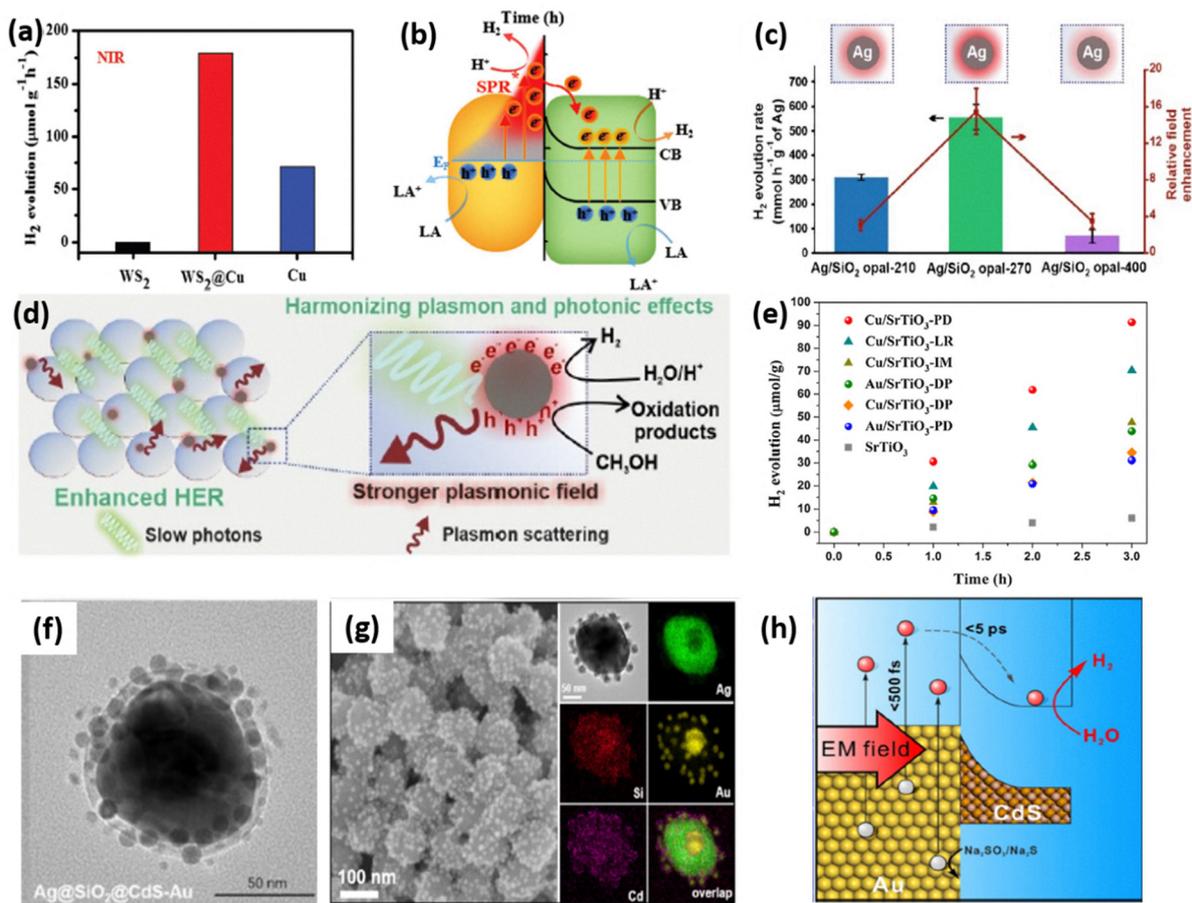


Fig. 3 (a) Schematic depiction of the Ag/TpPa-1-COF synthesis. (b) UV-vis DRS spectra of the as-synthesized photocatalysts. (c) Suggested mechanism for the release of H<sub>2</sub> by photocatalysis over Ag/TpPa-1. Reproduced with permission.<sup>54</sup> Copyright 2024, Royal Society of Chemistry. (d) Schematic illustration of the synthesis of Cu NPs/CdS/In<sub>2</sub>O<sub>3</sub>. (e) Photocatalytic H<sub>2</sub> evolution rate of Cu NPs/CdS/In<sub>2</sub>O<sub>3</sub> under light irradiation at different wavelengths. (f) Mechanism over the photocatalyst Cu NPs/CdS/In<sub>2</sub>O<sub>3</sub> for H<sub>2</sub> evolution coupled with selective FFA oxidation to FF. Reproduced with permission.<sup>55</sup> Copyright 2024, American Chemical Society.

deposition method. The XPS profile from the Cu 2p region showed two peaks at 932.7 and 952.5 eV for Cu 2p<sub>3/2</sub> and Cu 2p<sub>1/2</sub> that are so close to Cu<sup>+</sup> and Cu<sup>0</sup> states, which was further confirmed by Cu LMM XAES analysis identifying the presence of Cu<sup>0</sup> state. The WS<sub>2</sub>@Cu photocatalyst under irradiation with NIR light exhibited an average H<sub>2</sub> evolution rate of around 178.6 μmol g<sup>-1</sup> h<sup>-1</sup>, while bare Cu NPs showed a H<sub>2</sub> evolution rate of 71.4 μmol g<sup>-1</sup> h<sup>-1</sup>. Under NIR irradiation, WS<sub>2</sub> exhibited

no noticeable H<sub>2</sub> evolution (Fig. 4a). This result suggested that NIR irradiation generated hot electrons that were injected to the CB of WS<sub>2</sub> from plasmonic Cu over the Schottky barrier, facilitating the transfer of hot electrons to WS<sub>2</sub> and suppressing the recombination rate in Cu NPs, thereby enhancing the photocatalytic activity (Fig. 4b). In addition, the H<sub>2</sub> evolution activity further increases upon UV-vis light illumination as WS<sub>2</sub> can also generate photoexcited electrons in the CB in addition





**Fig. 4** (a) Photocatalytic  $\text{H}_2$  evolution on photocatalysts upon irradiation of  $>750$  nm NIR light. (b) Proposed schematic representation of photoinduced charge migration in  $\text{WS}_2@Cu$  for photocatalytic  $\text{H}_2$  evolution in the presence of sacrificial agent LA. Reproduced with permission.<sup>56</sup> Copyright 2018, Wiley. (c)  $\text{H}_2$  evolution and associated field enhancement across diverse plasmonic photonic crystals shown, with the inset displaying electromagnetic field strengths for different plasmonic photocatalysts, in agreement with earlier SERS analyses. (d) The superior light absorption of  $Ag/SiO_2$  opal-270 for improved  $\text{H}_2$  evolution. Reproduced with permission.<sup>57</sup> Copyright 2024, Wiley. (e)  $\text{H}_2$  evolution as a function of irradiation time under visible light ( $\lambda > 400$  nm) for 0.5 wt%  $Cu/SrTiO_3$  and  $Au/SrTiO_3$ . Reproduced with permission.<sup>58</sup> Copyright 2019, Elsevier. (f) HRTEM image of  $Ag@SiO_2@CdS-Au$ . Reproduced with permission.<sup>59</sup> Copyright 2020, American Chemical Society. (g) SEM image of  $Ag$  SHIN-Au@CdS nanoparticles. The inset illustrates a TEM image and the associated EDS mapping analysis of an individual  $Ag$  SHIN-Au@CdS nanoparticle. (h) Schematic illustration of the mechanism for enhancement in  $\text{H}_2$  evolution over  $Ag$  SHIN-Au@CdS dual antennas. Reproduced with permission.<sup>60</sup> Copyright 2021, American Chemical Society.

to hot electrons and thus the increase in  $\text{H}_2$  evolution activity is due to the synergistic effect of two optical excitations.<sup>56</sup>

Mogan and co-workers unprecedentedly designed a plasmonic photonic crystal by taking a non-photo absorbing photonic counterpart to intensify the plasmonic field by enhancing the slow photon generation to promote light-to-chemical transformation. The deposited  $Ag$  NPs over  $SiO_2$  minimized the excessive competition between plasmonic metals on the photonic effect, mainly by mitigating the adverse effects of random plasmonic light scattering and optical screening. The plasmonic photonic catalyst achieved an outstanding  $\text{H}_2$  evolution rate outperforming other high-performance hybrid photocatalysts by up to 280-fold and surpassing bare  $Ag$  nanoparticles by  $>106$ -fold (Fig. 4c). This is attributed to the effective trapping and concentration of light within the plasmonic photonic crystals that enhance the photo absorption by the crystal and intensify the local plasmonic field on the  $Ag$  catalyst *via* LSPR (Fig. 4d). A surplus of hot carriers was produced as a result of

non-radiative decay of the surface plasmons which promotes  $\text{H}_2$  evolution.<sup>57</sup>

To increase the lifetime of the hot electrons, various strategies like the use of heterojunctions, adjusting the reaction environment, and cocatalyst integration have been adopted, with metal particle size modulation being frequently overlooked. In this regard, Zhang *et al.* examined the effect of  $Cu$  NP size on the photocatalytic production of  $\text{H}_2$  by depositing  $Cu$  NPs ranging in size from 2.8 nm to 7.7 nm on the  $SrTiO_3$  support. The maximum detectable  $\text{H}_2$  generation activity of  $76.3 \mu\text{mol g}^{-1} \text{h}^{-1}$  was noted at a 0.5 weight percent  $Cu$  loading and an average diameter of 3.9 nm, which is five times more than that of  $Au/SrTiO_3$  (Fig. 4e). Larger  $Cu$  particle sizes were responsible for the increment in photocatalytic  $\text{H}_2$  generation activity because they strengthen the SPR effect and lengthen the carrier lifespan. It is crucial to remember that  $Cu$ -based catalyst stability is often low, and further study is required to find solutions.<sup>58</sup>



Ren and co-workers developed a core-shell-satellite-type plasmonic nanocomposite photocatalyst, Ag@SiO<sub>2</sub>@CdS-Au, with varying core and shell thicknesses by a seeded growth method (Fig. 4f). They validated the significant impact of near-field improvement and PRET mechanisms in enhancing photocatalysis by comparing the reactivity towards photocatalytic H<sub>2</sub> evolution. The inclusion of gold satellite nanoparticles helped to introduce hot electrons, expanding the range of light absorption for the photocatalyst and enhanced the electromagnetic field through Ag-Au coupling. The experimental findings demonstrated a significant improvement (approximately 200 times) in the efficiency of the photocatalytic hydrogen production process. The AQY spectra along with UV-vis spectra of the Ag@SiO<sub>2</sub>@CdS-Au core-shell-satellite nanocomposite and Ag@SiO<sub>2</sub>@CdS NPs were compared and it was observed that the Ag@SiO<sub>2</sub>@CdS-Au nanocomposite presented strong light absorption in the longer wavelength region (500-800 nm) due to the loading of Au satellite nanoparticles, outperforming Ag@SiO<sub>2</sub>@CdS NPs. The AQY plot of Ag@SiO<sub>2</sub>@CdS-Au displayed a similar trend but with much stronger light absorption in the short wavelength region (<500 nm) compared to Ag@SiO<sub>2</sub>@CdS.<sup>59</sup>

Another plasmon-mediated enhanced H<sub>2</sub> evolution performance was reported by Yang *et al.* using a dual-plasmonic-antenna that efficiently generated energetic hot electrons and strong electromagnetic fields simultaneously as shown in Fig. 4g. Briefly, they fabricated the dual plasmonic catalyst by assembling Au@CdS with a pinhole on Ag@SiO<sub>2</sub> and the H<sub>2</sub> evolution rate by the photocatalyst was improved by almost 7000 times compared with CdS. The dual-plasmonic-antenna nanocomposites demonstrate superior photocatalytic activity for H<sub>2</sub> evolution reaction under visible light irradiation. This is attributed to the efficient generation of a strong magnetic field and energetic hot electrons. The study shows that the hot electrons are produced in the Au cores of the Au@CdS satellites and then transferred to the conduction band of CdS, initiating the photocatalytic H<sub>2</sub> evolution. Additionally, the strong electromagnetic field generated by the Ag SHINs promotes the production and migration of hot electrons, providing additional uplifting in the overall activity (Fig. 4h).<sup>60</sup>

### 3.3. Bimetallic alloys as co-catalysts

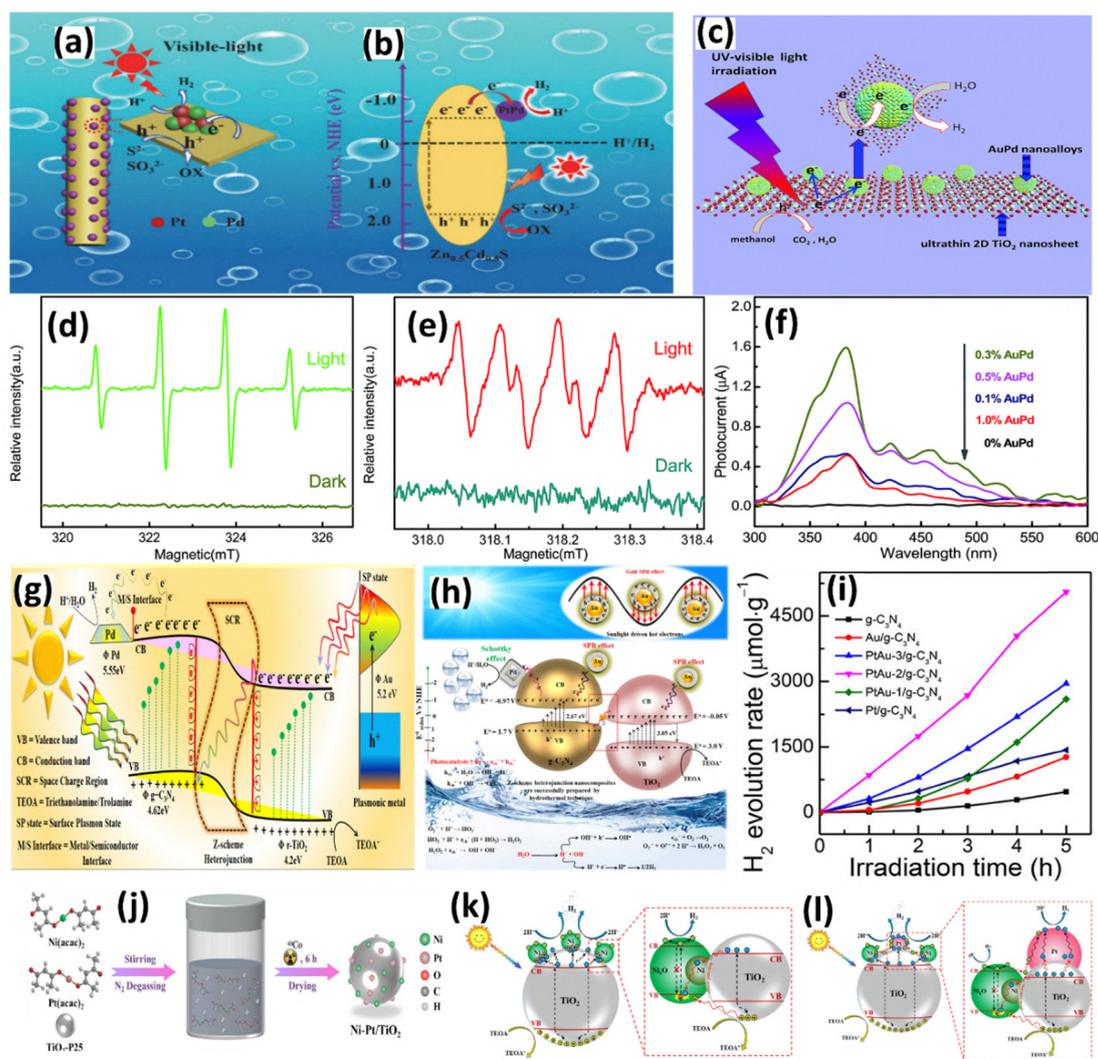
In recent years, bimetallic alloy nanoparticles (NPs) have increasingly drawn attention as cocatalysts for the photocatalytic generation of hydrogen from water. Studies have demonstrated that alloying a metal catalyst with a second metal can restrict the poisoning of the metal catalyst while enhancing its activity and selectivity.<sup>61</sup> In accordance with the fundamental principles governing photocatalytic H<sub>2</sub> production from H<sub>2</sub>O, bimetallic cocatalysts are anticipated to enhance the separation and migration of photogenerated excitons while providing adequate active sites for H<sub>2</sub> evolution. Given the vast array of elements in the periodic table, countless bimetallic compositions are possible when selecting two metals at random. However, only a limited number of these compositions exhibit remarkable synergistic effects, leading to superior co-catalytic

performance. As interest in bimetallic cocatalysts grows across various catalytic applications, significant advancements have been made in the development of bimetallic combinations, including bi-noble metals, noble-non-noble bimetallics, and bi-non-noble metal alloy cocatalysts for the light-driven reduction of water to H<sub>2</sub>. Noble metals from the periodic table, including platinum (Pt), palladium (Pd), gold (Au), and silver (Ag), are frequently employed as highly effective cocatalysts in various reduction reactions due to their low Fermi levels. These metals trap photogenerated electrons and facilitate the generation of a Schottky barrier at the interface of the metal and semiconductor. This mechanism effectively inhibits the rapid recombination of photo-induced charge carriers, thereby significantly enhancing the overall performance of photocatalytic processes.<sup>62</sup>

Zhang *et al.*<sup>63</sup> synthesized Pt-Pd bimetal NPs by a dual-step chemical reduction method and Zn<sub>0.5</sub>Cd<sub>0.5</sub>S nanorods were synthesized utilizing the solvent thermal method. Additionally, platinum-palladium (Pt-Pd) was uniformly deposited over the exterior surface of Zn<sub>0.5</sub>Cd<sub>0.5</sub>S through an *in situ* synthesis process and used in H<sub>2</sub> evolution from light-driven water-splitting. They found a significant improvement in the H<sub>2</sub> evolution rate; under visible light illumination. The PtPd/Zn<sub>0.5</sub>Cd<sub>0.5</sub>S nanorods demonstrated an optimum rate of H<sub>2</sub> production. Additionally, these nanorods achieved a notable apparent quantum yield (AQY) efficiency of up to 10.43% at a wavelength of 420 nm. The interface of PtPd NPs, characterized by high surface energy, functions as an active site for charge transfer. This characteristic facilitates improved photogenerated electron and hole separation, thereby enhancing the efficiency of the photocatalytic process and accelerating the overall rate of hydrogen evolution reaction (Fig. 5a and b). Furthermore, the synergistic effect of both the metals Pt and Pd within the alloy structure is evidenced by the superior photocatalytic activity of PtPd/Zn<sub>0.5</sub>Cd<sub>0.5</sub>S in comparison to Pt/Zn<sub>0.5</sub>Cd<sub>0.5</sub>S.

Xin *et al.*<sup>64</sup> synthesized AuPd bimetallic nanoalloy-decorated ultrathin two-dimensional titanium dioxide (TiO<sub>2</sub>) nanosheets *via an in situ* method for light-driven H<sub>2</sub> evolution from water using methanol as the sacrificial agent. 0.3 wt% AuPd/TiO<sub>2</sub> achieved an H<sub>2</sub> evolution rate of 526 μmol h<sup>-1</sup> g<sup>-1</sup>, surpassing the performance of pure TiO<sub>2</sub> by a factor of 31. The AuPd alloy nanoparticles, with their lower Fermi level, served as efficient electron traps, facilitating the transfer of photogenerated electrons from the conduction band of TiO<sub>2</sub> to the alloy surface (Fig. 5c). ESR and SPC analyses were employed to examine the roles of reactive species and photogenerated charge carriers. ESR results indicated the generation of superoxide (O<sub>2</sub><sup>•-</sup>) and hydroxyl radicals (•OH) under UV-vis light irradiation (Fig. 5d and e). The SPC analysis revealed that the 0.3 wt% AuPd/TiO<sub>2</sub> sample exhibited the highest signal intensity, indicating maximal charge carrier separation efficiency (Fig. 5f). However, excessive loading of AuPd nanoparticles reduced light absorption by TiO<sub>2</sub>, leading to diminished SPC signal intensity and lower photocatalytic performance. The findings underline the importance of optimizing bimetallic nanoparticle content to achieve maximum hydrogen production efficiency.





**Fig. 5** (a) and (b) Mechanism of  $\text{H}_2$  evolution on the PtPd/ZnCdS surface. Reproduced with permission.<sup>63</sup> Copyright 2021, Elsevier. (c) Illustration of the proposed mechanism for charge transfer and  $\text{H}_2$  evolution in the AuPd/TiO<sub>2</sub> composite under UV-visible light. (d) and (e) ESR spectra of AuPd/TiO<sub>2</sub> photocatalysts showing  $\text{O}_2^-$  detected in methanol and  $\cdot\text{OH}$  detected in water. (f) SPC spectra of the AuPd/TiO<sub>2</sub> photocatalyst with varying AuPd nanoparticle concentrations. Reproduced with permission.<sup>64</sup> Copyright 2015, Royal Society of Chemistry. (g) Graphical illustration of electron flow induced by the Au LSPR effect in the Z-scheme system ( $\text{g-C}_3\text{N}_4/\text{TiO}_2$ ), culminating in their transfer to the catalytic active sites of palladium (Pd) within the Pd/Au@g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> composite. (h) The mechanism of reaction involved in solar water splitting for the generation of hydrogen utilizing the Pd/Au@g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> composite. Reproduced with permission.<sup>65</sup> Copyright 2024, American Chemical Society. (i) The rate of hydrogen evolution was assessed for both pure g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub> integrated with metal and metal alloy nanoparticles. Reproduced with permission.<sup>66</sup> Copyright 2018, American Chemical Society. (j) Schematic representation of synthetic Ni–Pt/TiO<sub>2</sub> photocatalysts. (k) and (l) Proposed photocatalytic mechanisms for monometallic and bimetallic modifications over TiO<sub>2</sub>. Reproduced with permission.<sup>67</sup> Copyright 2023, American Chemical Society.

The quaternary system of Au–Pd NPs on TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> has also been studied for co-catalyzing photocatalytic  $\text{H}_2$  production from  $\text{H}_2\text{O}$ , using triethanolamine as a sacrificial agent to stabilize the system and enhance hydrogen production. Rauf and coworkers<sup>65</sup> successfully synthesized Pd/Au@g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> through a two-step process comprising chemical reduction, hydrothermal treatment, and subsequent calcination. The hydrothermal treatment effectively enabled the *in situ* incorporation of Pd/Au cocatalysts, and the  $\text{H}_2$  evolution was found to be 11.23- and 6.97-fold higher than that of pristine TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>, respectively. The integration of Pd–Au nanoparticles with the TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> heterostructures significantly promoted

electron transfer, attributed to the Z-scheme interaction between the components, as well as the localized surface plasmon resonance (LSPR) effect of Au and the Schottky barrier effect of Pd (Fig. 5g). This combination enhanced the photocatalytic hydrogen production by facilitating the separation of photogenerated electron–hole pairs and thus inhibiting recombination (Fig. 5h).

Bhunia *et al.*<sup>66</sup> conducted an analysis examining the performance of platinum–gold (Pt–Au) bimetallic nanoparticles (3–5 nm) incorporated onto g-C<sub>3</sub>N<sub>4</sub> at varying concentrations for the photocatalytic production of  $\text{H}_2$  from water, utilizing Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>S as sacrificial agents. The alloy nanoparticles



with a Pt: Au atomic ratio of 1 demonstrated the highest photocatalytic activity when supported on g-C<sub>3</sub>N<sub>4</sub>, in addition to favorable results in recyclability assessments (Fig. 5i). The reported H<sub>2</sub> evolution rate for bare g-C<sub>3</sub>N<sub>4</sub> was 94 μmol g<sup>-1</sup> h<sup>-1</sup>, while the PtAu-2/g-C<sub>3</sub>N<sub>4</sub> configuration exhibited a markedly elevated rate of 1009 μmol g<sup>-1</sup> h<sup>-1</sup>, reflecting an increase of nearly tenfold.

Considering the limited reserves and elevated costs associated with noble metals, significant efforts have been directed toward enhancing catalytic performance while minimizing the quantity of these metals utilized. Another notable strategy involves the alloying of two different categories of metals like noble metals with cost-effective and earth abundant non-noble transition metals, including Ni, Cu, and Ru. Notably, numerous noble-non-noble bimetal alloy NPs have demonstrated exceptional efficacy as co-catalysts, frequently surpassing the performance of pure noble metals in the photocatalytic evolution of H<sub>2</sub> from water.

The study by Wang *et al.*<sup>67</sup> investigated the photocatalytic H<sub>2</sub> evolution using Ni-Pt bimetallic cocatalysts supported on TiO<sub>2</sub> *via* radiolytic reduction. The authors synthesized ultrasmall (1.9 nm) Ni-Pt NPs with precise dispersion on the TiO<sub>2</sub> surface (Fig. 5j). Ni serves as an electron donor, thereby extending the lifetime of photogenerated charge carriers by providing additional electrons to the system. In contrast, Pt functions as an electron trap due to its elevated work function, effectively utilizing electrons for proton reduction. This electron transfer pathway is facilitated by the close contact between Ni-Pt NPs, which creates a multijunction structure on TiO<sub>2</sub>. In the context of monometallic systems, as illustrated in Fig. 5k, electrons migrate from TiO<sub>2</sub> to the metal nanoparticles, where they are subsequently captured for proton reduction, with Pt exhibiting superior activity owing to its enhanced electron affinity. Conversely, in bimetallic systems, as depicted in Fig. 5l, a distinct pathway is established: photogenerated electrons are initially donated by Ni and are then efficiently captured by Pt at the Ni-Pt interface. This mechanism significantly enhances electron utilization and suppresses recombination. Furthermore, optimization of the Ni-Pt ratio has revealed that the Ni<sub>2</sub>-Pt<sub>1</sub> configuration achieves the highest performance and an apparent quantum yield (AQY) of 77.5%, approximately 40 times greater than the performance of bare TiO<sub>2</sub>.

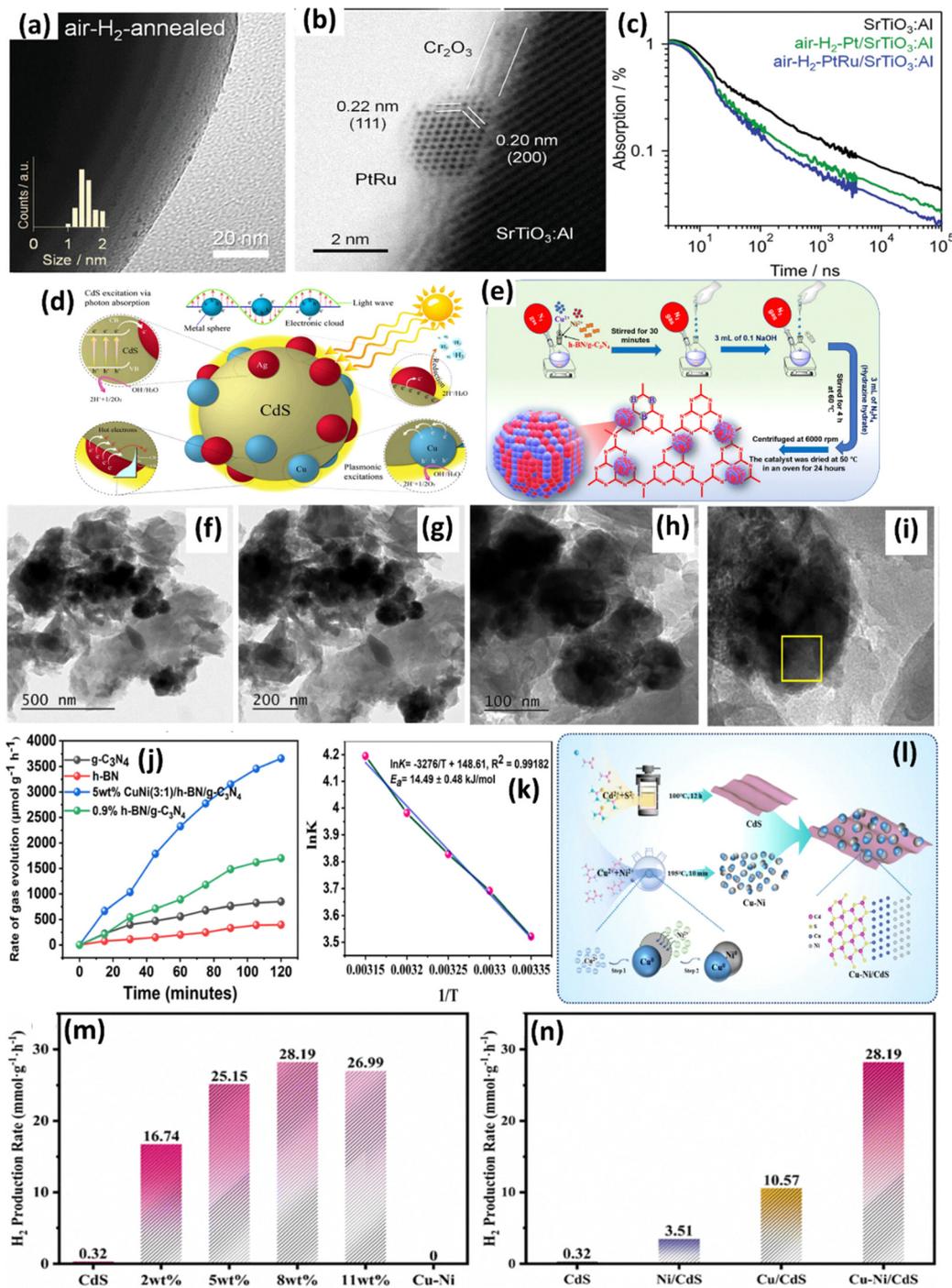
Pelicano *et al.*<sup>68</sup> conducted a comprehensive investigation into the photocatalytic H<sub>2</sub> evolution capabilities of ultrafine PtRu bimetallic alloy NPs supported on Al-doped SrTiO<sub>3</sub>. The study underscored the synergistic interactions between Pt and Ru atoms. Utilizing the application of a polyol method of synthesis, the researchers produced PtRu NPs with an average diameter of 1.0 ± 0.2 nm, which were uniformly deposited onto SrTiO<sub>3</sub>:Al *via* liquid-phase adsorption, followed by annealing in both air and hydrogen. From a mechanistic perspective, the inclusion of Ru within the Pt lattice is shown to reduce the work function of Pt, as demonstrated in Fig. 6a and b. This modification results in a decreased Schottky barrier height, thereby enhancing the transfer of photoexcited electrons from SrTiO<sub>3</sub>:Al to the cocatalyst. Data obtained from transient diffuse

reflectance spectroscopy, as illustrated in Fig. 6c, confirm that the PtRu/SrTiO<sub>3</sub>:Al catalyst facilitates a more rapid decay of photoexcited electrons, achieving an apparent quantum yield of 65% at 365 nm, which exceeds the performance of monometallic Pt and Ru catalysts. Furthermore, the catalyst demonstrates exceptional stability, maintaining its activity for more than 20 hours without degradation. This work positions PtRu alloys as cost-effective alternatives to conventional noble-metal-only cocatalysts, highlighting their promising potential for scalable photocatalytic H<sub>2</sub> production.

Hussain *et al.*<sup>69</sup> conducted a thorough investigation into the application of a Cu-Ag cocatalyst on CdS for the generation of hydrogen through sunlight-driven processes. The cocatalyst was synthesized through a hydrothermal method, during which Cu and Ag were chemically reduced and deposited onto the surface of CdS. This process resulted in the formulation Cu<sub>2%</sub>-Ag<sub>1%</sub>/CdS, which exhibited a notable hydrogen evolution rate with a quantum efficiency of 45.04% at 420 nm. Upon exposure to sunlight, CdS absorbs photons, thereby exciting electrons from the VB to the CB, generating holes in the VB. The surface plasmon resonance properties of Cu facilitate the generation of hot electrons that migrate to the CB of CdS, thereby enhancing the electron density at the surface of the semiconductor. Concurrently, Ag forms Schottky junctions at the CdS interface, functioning as electron sinks that efficiently capture photogenerated electrons and mitigate their recombination with holes (Fig. 6d). This research illustrated the significant potential of utilizing bimetallic cocatalysts in addressing the challenges associated with photocorrosion and increasing the reusability of the catalyst thereby enhancing the efficiency of hydrogen evolution under sunlight exposure.

To achieve cost-effective, competent, and durable cocatalysts for photocatalytic H<sub>2</sub> evolution, considerable attention has been directed toward low-cost non-noble transition metals, including Ni, Cu, and Ru. Nevertheless, the performance of individual non-noble transition metal cocatalysts remains limited, and a propensity for oxidation compromises their long-term stability. The strategic combination of these metals may facilitate the optimization of their properties, potentially resulting in a level of performance that exceeds that of noble metals. The CuNi alloy nanoparticles supported on h-BN/g-C<sub>3</sub>N<sub>4</sub> were synthesized using a surfactant-free deposition-precipitation method (Fig. 6e), ensuring a uniform distribution of nanoparticles. Structural analysis *via* XPS and XANES confirmed the metallic state of the nanoparticles. At the same time, TEM revealed well-dispersed CuNi alloy nanoparticles with an average size of 90–100 nm anchored onto the h-BN/g-C<sub>3</sub>N<sub>4</sub> nanosheets (Fig. 6f–i). The 5 wt% CuNi (3:1)/h-BN/g-C<sub>3</sub>N<sub>4</sub> nanocomposite exhibited the highest photocatalytic hydrogen evolution rate of 3658.9 μmol g<sup>-1</sup> h<sup>-1</sup>, outperforming pristine g-C<sub>3</sub>N<sub>4</sub> and CuNi (3:1) nanoparticles by 4.31 and 1.82 times, respectively (Fig. 6j). This enhanced activity is attributed to the synergistic effects of Cu and Ni, with Cu enhancing visible light absorption through its localized surface plasmon resonance and Ni serving as an electron sink to suppress charge recombination. The optimization of CuNi composition and decoration





**Fig. 6** (a) Air-H<sub>2</sub>-PtRu/SrTiO<sub>3</sub>:Al. (b) BF-STEM. (c) The time profile of the TDR signal measured at 0.48–0.67 eV following a 3.54 eV pump pulse for SrTiO<sub>3</sub>:Al, with and without the cocatalysts (pump fluence was 35  $\mu\text{J}$  per pulse, resulting in a photon density of  $2.21 \times 10^{18}$  photons  $\text{cm}^{-3}$ , Cr<sub>2</sub>O<sub>3</sub> was used as a cocatalyst to prevent oxidation). Reproduced with permission.<sup>68</sup> Copyright 2022, John Wiley and Sons. (d) Mechanism of photocatalytic hydrogen production utilizing Cu<sub>2</sub>/Ag<sub>1</sub>@CdS catalysts. Reproduced with permission.<sup>69</sup> Copyright 2024, American Chemical Society. (e) The synthesis of a 5 wt% CuNi (3 : 1)/h-BN/g-C<sub>3</sub>N<sub>4</sub> nanocomposite. (f)–(i) Transmission electron microscopy (TEM) images of a 5 wt% CuNi (3 : 1)/h-BN/g-C<sub>3</sub>N<sub>4</sub> nanocomposite at various magnifications. (j) The rates of hydrogen gas evolution were investigated for various materials, including g-C<sub>3</sub>N<sub>4</sub>, h-BN, a composite of 0.9% h-BN/g-C<sub>3</sub>N<sub>4</sub>, and a formulation of 5 wt% CuNi(3 : 1)/h-BN/g-C<sub>3</sub>N<sub>4</sub>, under conditions that included the presence of a scavenger. (k) Arrhenius plots serving as the method for defining the activation energy ( $E_a$ ) of chemical reactions. Reproduced with permission.<sup>70</sup> Copyright 2024, American Chemical Society. (l) Depiction of Cu–Ni/CdS hybrid fabrication. (m) and (n) Comparative analysis of the H<sub>2</sub> generation rates for various compositions of Cu–Ni and a range of different metals. Reproduced with permission.<sup>71</sup> Copyright 2024, Royal Society of Chemistry.



on the h-BN/g-C<sub>3</sub>N<sub>4</sub> matrix significantly improved the material's hydrophilicity, charge mobility, and photocatalytic efficiency. Additionally, the temperature-dependent study demonstrated that increasing the reaction temperature to 45 °C reduced the activation energy to 14.49 kJ mol<sup>-1</sup> (Fig. 6k), further boosting hydrogen production. This study highlights the potential of CuNi/h-BN/g-C<sub>3</sub>N<sub>4</sub> composites for sustainable and efficient solar-driven hydrogen evolution.<sup>70</sup>

Zeng *et al.*<sup>71</sup> developed a ternary photocatalyst system that integrates plasmonic Cu–Ni bimetal nanoparticles with ultrathin CdS nanosheets through a solution-phase synthesis method (Fig. 6l). The optimized 8% Cu–Ni/CdS nanocomposite achieved an outstanding photocatalytic hydrogen generation under visible light, representing an improvement of 88 times compared to pure CdS, and it significantly outperformed its monometallic counterparts, as illustrated in Fig. 6m. Moreover, the composite exhibited a high apparent quantum yield (AQY) of 21.5% at 400 nm and 12.1% at 520 nm, indicating its efficiency across a wide range of the light spectrum. Fig. 6n further demonstrates that the performance of the ternary composite exceeded that of Cu/CdS and Ni/CdS. This enhancement can be attributed to the synergistic effects of Cu and Ni, where Cu improves visible light absorption through surface plasmon resonance (SPR), while Ni establishes Schottky junctions that effectively reduce charge recombination. This study highlights the potential of non-noble bimetallic cocatalysts in facilitating cost-effective solar energy conversion. However, structural stability of the bimetallic alloy cocatalysts is often compromised under photocatalytic operating conditions where selective leaching, phase segregation, or oxidation of one component can disrupt the alloyed state and reduce activity.

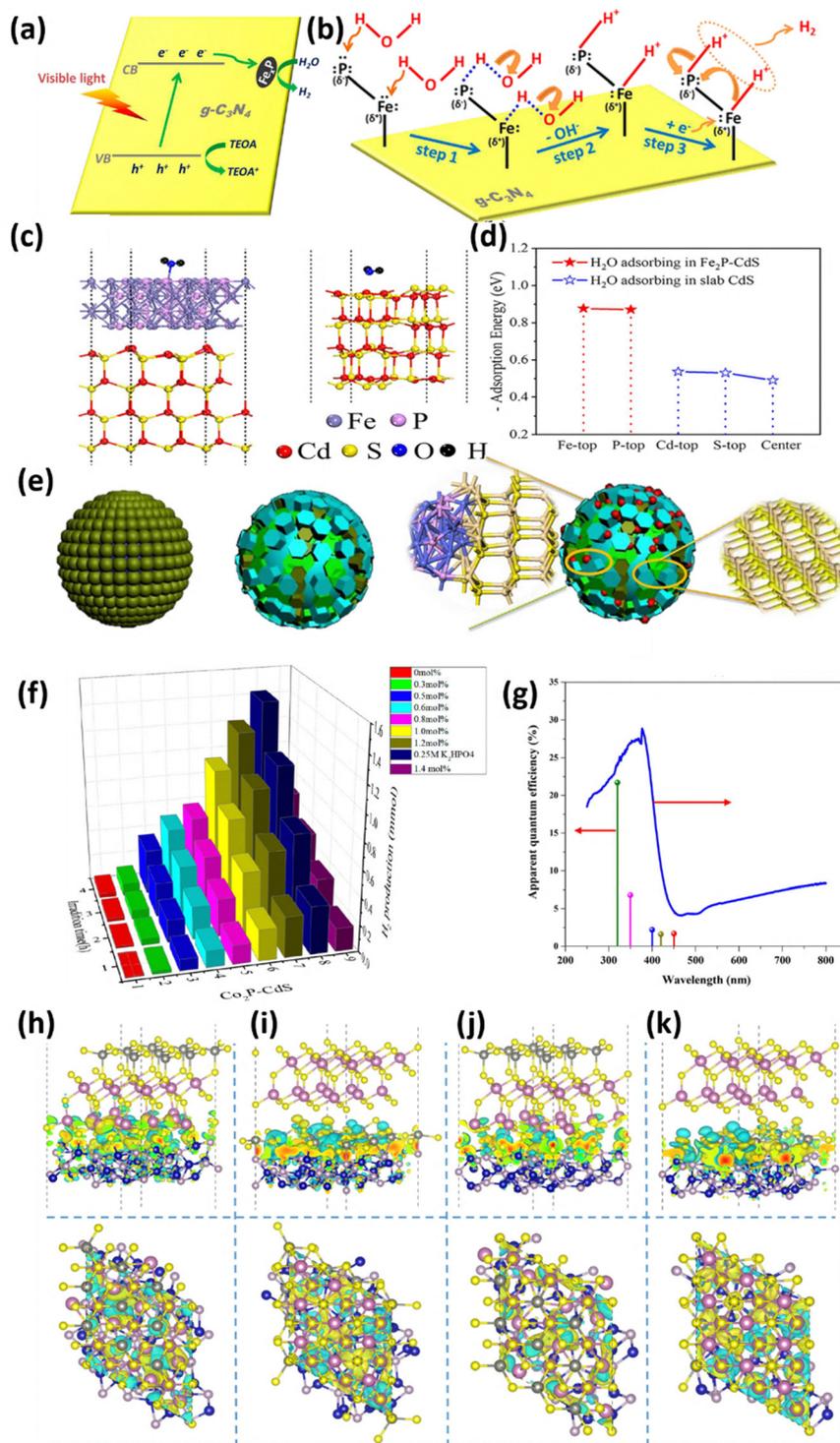
### 3.4. Metal phosphides as co-catalysts

Metal phosphides are promising materials for energy storage and conversion due to their distinctive properties, including exceptional electrical conductivity, catalytic activity, and chemical stability. Their application as photocatalysts and cocatalysts for hydrogen generation has garnered considerable attention in recent research. Metal phosphides based on elements such as Ni, Fe, Co, Cu, Mo, W, Ru, and Rh are particularly noteworthy. These materials typically exhibit metallic characteristics with work function values in the range of 4.5–5.1 eV, facilitating the effective capture and transfer of photoinduced electrons from semiconductor photocatalysts to the reaction sites. The metallic nature of these phosphides ensures low resistance to electron transport, which minimizes charge recombination within the photocatalyst and improves overall catalytic efficiency. Moreover, metal phosphides offer abundant active sites for hydrogen evolution, attributable to their surface composition and electronic structure, which promote proton reduction. However, establishment of a Schottky junction at the metal phosphide and semiconductor photocatalyst interface promotes band bending and prevents back flow of electrons which further enhances charge separation, thereby extending the lifetime of photogenerated electrons and holes.

The development and study of non-precious metal co-catalysts are crucial for achieving efficient photocatalytic hydrogen production. Iron due to its abundance and low cost is a promising material for iron-based co-catalysts for photocatalytic hydrogen evolution. Zhao *et al.*<sup>72</sup> have introduced a novel g-C<sub>3</sub>N<sub>4</sub>/Fe<sub>x</sub>P hybrid photocatalyst intended for visible-light-driven hydrogen evolution. This innovative photocatalyst integrates g-C<sub>3</sub>N<sub>4</sub> with a noble-metal-free iron phosphide (Fe<sub>x</sub>P) cocatalyst. The synthesis of the hybrid occurs through a two-step process: initially, g-C<sub>3</sub>N<sub>4</sub>/FeOOH is prepared *via* hydrothermal methods, followed by phosphidation using sodium hypophosphite at 300 °C in an argon environment. The g-C<sub>3</sub>N<sub>4</sub>/Fe<sub>x</sub>P hybrid demonstrated remarkable photocatalytic performance, achieving a hydrogen evolution rate that is 277 fold greater than that of pristine g-C<sub>3</sub>N<sub>4</sub>, which can be comparable to systems loaded with platinum. This substantial enhancement is attributed to the synergistic dual-site mechanism present in Fe<sub>x</sub>P, where Fe(δ<sup>+</sup>) and P(δ<sup>-</sup>) atoms serve as centers for hydride and proton adsorption, respectively. This configuration facilitates efficient charge separation and promotes water reduction. As depicted in Fig. 7a and b, photo-generated electrons are transferred from g-C<sub>3</sub>N<sub>4</sub> to Fe<sub>x</sub>P, thereby accelerating the hydrogen evolution process. Concurrently, the sacrificial agent (TEOA) functions to scavenge holes, reducing the likelihood of recombination. This robust interaction not only ensures high stability and efficiency but also presents a cost-effective and sustainable alternative to noble-metal-based photocatalysts for hydrogen production.

Liang *et al.*<sup>73</sup> advanced the application of metal phosphides in the realm of photocatalysis by presenting a highly efficient 2D–1D Fe<sub>2</sub>P–CdS heterojunction photocatalyst. This innovative composite is designed to address the inherent challenges associated with CdS nanorods, notably rapid electron-hole recombination and insufficient stability. The Fe<sub>2</sub>P nanosheets, which are synthesized through a mild solvothermal method, exhibit an ultrathin structure (~1.41 nm) characterized by an abundance of defect sites that facilitate exceptional catalytic activity. The integration of Fe<sub>2</sub>P nanosheets with CdS nanorods results in the formation of robust heterojunctions, which significantly augment charge transfer and separation efficiency. This composite has demonstrated a remarkable H<sub>2</sub> evolution under visible light, surpassing the performance of bare CdS by a factor of 78.7. The superior catalytic performance of the Fe<sub>2</sub>P–CdS composite can be attributed to several synergistic mechanisms. Notably, a strong ionic interaction between Fe<sub>2</sub>P and CdS, as established by charge density difference calculations, promotes the migration of electrons from CdS to Fe<sub>2</sub>P, thereby mitigating recombination losses. Furthermore, as evidenced in Fig. 7c and d, the presence of Fe<sub>2</sub>P significantly enhances water adsorption, with calculated adsorption energy values ranging from –0.870 to –0.876 eV, which exceed those of pristine CdS (–0.490 to –0.536 eV). This enhanced water adsorption, in combination with a lower hydrogen reduction overpotential, serves to accelerate the hydrogen evolution. Employing experimental techniques such as time-resolved photoluminescence and electron spin resonance, it has been





**Fig. 7** (a) Diagrammatic representation of the photocatalytic  $\text{H}_2$  evolution over  $g\text{-C}_3\text{N}_4/\text{Fe}_x\text{P}$ -0.08 when exposed to visible light. (b) Schematic representation of charge transfer and reaction mechanism for the enhanced  $\text{H}_2$  evolution by the  $\text{Fe}_x\text{P}$  cocatalyst on the  $g\text{-C}_3\text{N}_4$  surface. Reproduced with permission.<sup>72</sup> Copyright 2017, American Chemical Society. (c)  $\text{H}_2\text{O}$  adsorbed structures on the  $\text{Fe}_2\text{P}$ -CdS and slab CdS surfaces. Reproduced with permission.<sup>73</sup> Copyright 2021, Elsevier. (d)  $\text{H}_2\text{O}$  adsorption energy at various  $\text{Fe}_2\text{P}$ -CdS and slab CdS sites. Reproduced with permission.<sup>73</sup> Copyright 2021, Elsevier. (e) Method for creating cauliflower-shaped CdS- $\text{Co}_2\text{P}$  sub-microspheres. (f) Under visible light, photocatalytic  $\text{H}_2$  evolution across H- $\text{Co}_2\text{P}$ -CdS composite samples with varying  $\text{Co}_2\text{P}$  concentrations. Reproduced with permission.<sup>75</sup> Copyright 2018, American Chemical Society. (g) Apparent quantum efficiency over 10%  $\text{Co}_2\text{P}/\text{ZnIn}_2\text{S}_4$  NCGs. (h)–(k) Front and top images of the interface between (001)  $\text{Co}_2\text{P}$  and (001)  $\text{ZnIn}_2\text{S}_4$  showing four atomically matched interface topologies and electron density distributions. S, In, Zn, P, and Co atoms are represented by the yellow, purple, gray, brown, and blue balls, respectively. The accumulated and insufficient electron density is shown by the yellow and green area. Reproduced with permission.<sup>76</sup> Copyright 2021, Elsevier.



demonstrated that the 2D–1D heterostructure effectively suppresses charge recombination, resulting in the generation of abundant photo-reactive species. Additionally, the Fe<sub>2</sub>P–CdS hybrid exhibits notable stability over repeated cycles, underscoring its potential as a sustainable, noble-metal-free photocatalyst for H<sub>2</sub> evolution.

The promising performance of iron phosphide-based systems, such as FeP/CdS and FeP/g-C<sub>3</sub>N<sub>4</sub>, has been further enhanced by the incorporation of nickel, as demonstrated by Zhu *et al.*<sup>74</sup> This development led to the formation of NiFeP/g-C<sub>3</sub>N<sub>4</sub>, which exhibits superior photocatalytic activity for H<sub>2</sub> evolution. The synthesis of this system was achieved through a controlled phosphidation process of NiFe(OH)<sub>2</sub> supported on g-C<sub>3</sub>N<sub>4</sub>. The optimized NiFeP/g-C<sub>3</sub>N<sub>4</sub> configuration achieved a remarkable H<sub>2</sub> evolution under visible light, significantly outperforming single-metal phosphides. This enhancement is ascribed to the synergistic interactions between nickel and iron, which promote directional electron migration and lower the energy barriers for H<sub>2</sub> evolution. Evaluations utilizing photocurrent and electrochemical impedance spectroscopy indicated improved charge separation and transfer efficiency. Furthermore, fluorescence lifetime measurements underscored effective electron localization at the NiFeP surface. The system displayed considerable stability, maintaining its activity over multiple cycles, and exhibited an apparent quantum yield of 4.98% at 420 nm. Mechanistic insights revealed the efficient redirection of photogenerated electrons towards the NiFeP sites, while holes on g-C<sub>3</sub>N<sub>4</sub> facilitated water decomposition. These conclusions emphasize the significant potential of bimetallic phosphides as co-catalysts for sustainable H<sub>2</sub> evolution.

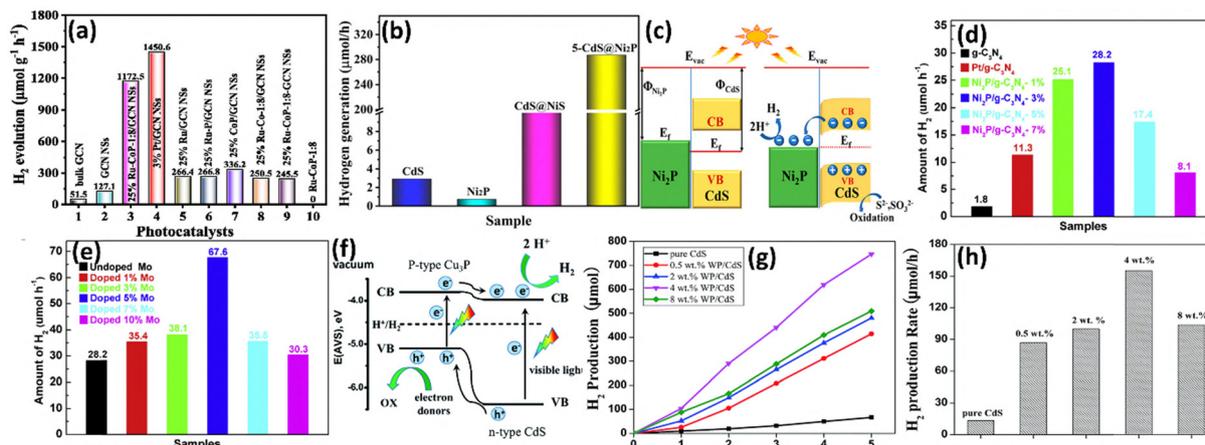
The study by Li *et al.*<sup>75</sup> unveils the remarkable potential of Co<sub>2</sub>P–CdS nanohybrids as a non-noble-metal based photocatalyst, setting a new benchmark for H<sub>2</sub> evolution. The photocatalyst was synthesized *via* an innovative *in situ* hydrothermal method; Co<sub>2</sub>P nanoparticles were seamlessly integrated onto CdS sub-microspheres, forming a unique cauliflower-like structure (Fig. 7e) that enhances light absorption and provides abundant active sites for catalytic activity. This strong interfacial bonding facilitates efficient charge transfer while minimizing recombination losses, driving the optimized nanohybrid (1.2 mol% Co<sub>2</sub>P) to achieve a remarkable H<sub>2</sub> evolution rate, outperforming Pt-loaded CdS by threefold and surpassing pure CdS by 41 times. The performance further soared with the addition of K<sub>2</sub>HPO<sub>4</sub> as a sacrificial agent, as illustrated by the time-dependent hydrogen production profile (Fig. 7f), highlighting the catalyst's stability and prolonged efficiency under visible-light irradiation. This synergistic interplay of tailored structure and enhanced charge separation establishes Co<sub>2</sub>P–CdS as a trailblazer in sustainable H<sub>2</sub> production, redefining the role of non-noble-metal materials in photocatalysis.

In another study, Zhang *et al.*<sup>76</sup> developed a hierarchical photocatalyst by coating hollow cobalt Co<sub>2</sub>P nanocages with a thin layer of ZnIn<sub>2</sub>S<sub>4</sub>, offering a highly efficient and stable noble-metal-free solution for photocatalytic H<sub>2</sub> evolution. The Co<sub>2</sub>P nanocages, derived from ZIF-67 precursors through a calcination and phosphiding process in a nitrogen atmosphere,

feature a hollow structure that enhances the surface area and facilitates electron diffusion. The subsequent hydrothermal deposition of ZnIn<sub>2</sub>S<sub>4</sub> forms a closely integrated heterojunction, significantly improving interfacial charge transfer and reducing electron–hole recombination. The interfacial charge migration and electronic structure of the (001) Co<sub>2</sub>P and (001) ZnIn<sub>2</sub>S<sub>4</sub> heterojunction were investigated using DFT calculations (Fig. 7h–k). These simulations revealed that electrons accumulate on the Co<sub>2</sub>P surface, while ZnIn<sub>2</sub>S<sub>4</sub> exhibits electron deficiency, indicating a robust electron transfer from ZnIn<sub>2</sub>S<sub>4</sub> to Co<sub>2</sub>P at the heterostructure interface. This electron migration aligns with XPS findings, highlighting the robust interfacial interaction that facilitates the efficient separation of photoexcited charge carriers. The hollow nanocage structure of Co<sub>2</sub>P further enhances this effect by providing an abundant surface area for charge transfer, ultimately boosting photocatalytic activity. The photocatalyst achieved an exceptional hydrogen evolution rate under simulated sunlight, which was approximately 10 times higher than that of pristine ZnIn<sub>2</sub>S<sub>4</sub>, and surpassed the performance of 1% noble-metal-loaded ZnIn<sub>2</sub>S<sub>4</sub>. With an apparent quantum efficiency of 21.7% at 320 nm, the Co<sub>2</sub>P/ZnIn<sub>2</sub>S<sub>4</sub> nanocages demonstrated superior activity and stability, maintaining performance over multiple cycles with minimal structural degradation (Fig. 7g). This work not only highlights the potential of transition metal phosphides like Co<sub>2</sub>P as effective noble-metal-free cocatalysts but also provides mechanistic insights into the interfacial electron dynamics, establishing a sustainable and cost-effective approach for hydrogen production.

Bimetallic systems such as cobalt phosphorus decorated with MCo<sub>2</sub>P (M = Ru, Ni) as a cocatalyst loaded on g-C<sub>3</sub>N<sub>4</sub> can be a good alternative to Pt/g-C<sub>3</sub>N<sub>4</sub> catalysts. In this regard, Meng *et al.*<sup>77</sup> and Li *et al.*<sup>78</sup> separately studied Ru-CoP/g-C<sub>3</sub>N<sub>4</sub> and NiCoP/g-C<sub>3</sub>N<sub>4</sub> based catalysts, respectively, contributing to significant advancements in designing cost-effective and highly efficient systems for photocatalytic H<sub>2</sub> evolution. In the Ru-CoP/g-C<sub>3</sub>N<sub>4</sub> system, a 2D/2D assembly structure was employed, integrating Ru-modulated CoP nanosheets with g-C<sub>3</sub>N<sub>4</sub> nanosheets. This configuration achieved an outstanding H<sub>2</sub> evolution rate under visible light irradiation with an apparent quantum efficiency of 3.49% (~420 nm), nearer to the performance of Pt-based systems (Fig. 8a). The superior activity was attributed to the synergistic effects of Ru and CoP, which promoted efficient charge separation, accelerated surface water reduction kinetics, and created a strong Schottky interface effect. Similarly, the NiCoP/g-C<sub>3</sub>N<sub>4</sub> heterostructure demonstrated remarkable photocatalytic efficiency, with a hydrogen evolution rate 2.27 times higher than that of Pt/g-C<sub>3</sub>N<sub>4</sub> under comparable conditions. This system leveraged a mesoporous 3D/2D design, where NiCoP nanoclusters were anchored on g-C<sub>3</sub>N<sub>4</sub> nanosheets. The presence of dual Co–N and Ni–N bonding states facilitated enhanced charge transfer and separation while maintaining structural stability over prolonged cyclic tests. The apparent quantum efficiency reached 4.8% at 420 nm, and the catalyst demonstrated exceptional durability, sustaining stable activity over 60 hours. Both systems





**Fig. 8** (a) Comparison of the typical rate of hydrogen evolution for the photocatalysts over 5 hours. Reproduced with permission.<sup>77</sup> Copyright 2021, Elsevier. (b) The rate at which various catalysts produce hydrogen by photocatalysis. (c) Diagrammatic representation of CdS@Ni<sub>2</sub>P charge transport. Reproduced with permission.<sup>78</sup> Copyright 2022, American Chemical Society. (d) H<sub>2</sub> evolution rates over samples of Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> with varying Ni<sub>2</sub>P contents. (e) H<sub>2</sub> evolution rates over the Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub>-3% sample with varying Mo doping. Reproduced with permission.<sup>80</sup> Copyright 2020, Elsevier. (f) A potential reaction pathway for the Cu<sub>3</sub>P/CdS heterojunction-based photocatalytic evolution of H<sub>2</sub>. Reproduced with permission.<sup>82</sup> Copyright 2015, Royal Society of Chemistry. (g) and (h) The rate of production of hydrogen (g) with time and (h) over WP/CdS photocatalysts at different WP loading concentrations (0.5 wt%, 2.0 wt%, 4.0 wt%, and 8.0 wt%). Reproduced with permission.<sup>84</sup> Copyright 2017, Royal Society of Chemistry.

underscore the potential of transition metal phosphides as noble-metal-free alternatives in photocatalysis, with tailored heterostructures and interfacial bonding playing crucial roles in optimizing performance. These studies provide valuable insights for developing advanced photocatalysts that combine high activity, stability, and scalability for sustainable hydrogen production.

Wang's group<sup>79</sup> initially described Ni<sub>2</sub>P as a remarkably effective cocatalyst for visible light driven photocatalytic H<sub>2</sub> production. It is evident that by employing Ni<sub>2</sub>P/CdS as a cocatalyst-semiconductor system, a substantial improvement in photocatalytic H<sub>2</sub> evolution under visible light was achieved. Ni<sub>2</sub>P/CdS was synthesized by a hydrothermal method followed by an *in situ* photodeposition method. Optimized Ni<sub>2</sub>P/CdS exhibits an H<sub>2</sub> evolution rate of 287 μmol h<sup>-1</sup>, surpassing that of bare CdS by over 98.3-fold (Fig. 8b). This remarkable improvement can be ascribed to the suitable Fermi level alignment (Fig. 8c). Additionally, Mo-doped Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> is an approach to increase the photocatalytic H<sub>2</sub> evolution activity in visible light. Dong's group<sup>80</sup> reported that Mo-Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> surpassed Pt/g-C<sub>3</sub>N<sub>4</sub> and Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> by over 6- and 2.4-fold, respectively, with an evolution rate of 67.6 μmol h<sup>-1</sup> at an optimized 5% Mo doping level (Fig. 8d and e). This enhancement is mainly attributed to the synergistic effect charge transfer to the MoNi<sub>2</sub>P nanodots from the nanosheet of g-C<sub>3</sub>N<sub>4</sub> and hole scavenging by TEOA from the valence band. Molybdenum-based materials show a high photocatalytic H<sub>2</sub> evolution rate. Especially, the metallic nature of MoP enables electron capture and effectively enhances the migration of photo-induced charges. The study by Yue *et al.*<sup>81</sup> reported a first-time MoP cocatalyst synthesized by temperature-programmed reduction and doped on CdS for high photocatalytic H<sub>2</sub> activity, with a quantum yield of ~5.8% and an evolution rate of 163.2 μmol h<sup>-1</sup> mg<sup>-1</sup>, surpassing that of pure and calcined CdS by over 9- and 20-fold, respectively.

Copper phosphide (Cu<sub>3</sub>P), a p-type semiconductor, has been validated as an effective co-catalyst for numerous photocatalytic reactions such as CO<sub>2</sub> reduction and H<sub>2</sub> production when integrated with semiconductors like CdS, g-C<sub>3</sub>N<sub>4</sub>, TiO<sub>2</sub>, *etc.* A study by Sun *et al.*<sup>82</sup> first discovered Cu<sub>3</sub>P as an effective cocatalyst for H<sub>2</sub> evolution when loaded onto CdS by a solvothermal method. The combination of Cu<sub>3</sub>P and CdS forms a p-n junction, with Cu<sub>3</sub>P acting as a p-type junction and CdS as an n-type junction (Fig. 8f). Under visible light, a H<sub>2</sub> evolution rate of ~200 μmol h<sup>-1</sup> mg<sup>-1</sup> with a quantum yield of ~25% was achieved under optimized conditions. Further Zhou *et al.*<sup>83</sup> conducted research to demonstrate the remarkable photocatalytic H<sub>2</sub> production performance of g-C<sub>3</sub>N<sub>4</sub> loaded with Cu<sub>3</sub>P and Cu<sub>9</sub>P<sub>3</sub> synthesized by an *in situ* grinding and phosphorylation method. The photocatalytic H<sub>2</sub> evolution rate of Cu<sub>3</sub>P/g-C<sub>3</sub>N<sub>4</sub> and Cu<sub>9</sub>P<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> was reported to be 343 μmol h<sup>-1</sup> g<sup>-1</sup> and 162.9 μmol h<sup>-1</sup> g<sup>-1</sup>, respectively.

Based on the similarities between Rh and Co and between W and Mo as they present in the same group, some W and Rh phosphide-based co-catalysts were developed for photocatalytic H<sub>2</sub> evolution. In a 2017 study, Zhang *et al.*<sup>84</sup> synthesized WP NPs *via* temperature programming reduction and doped them on semiconductor CdS. Following optimization, the 4% doped WP co-catalyst on CdS exhibited a high photocatalytic H<sub>2</sub> evolution rate of 155.2 μmol h<sup>-1</sup>, surpassing that of pristine CdS by over 11.6-fold and being 1/3 times the rate of Pt/CdS (Fig. 8g and h). Chen *et al.*<sup>85</sup> synthesized single-site-like Rh-phosphide doped on g-C<sub>3</sub>N<sub>4</sub> *via* a hydrothermal method followed by phosphorylation. As indicated by DFT calculations of reaction coordination of RhP reduces the Gibbs free energy of g-C<sub>3</sub>N<sub>4</sub>, resulting in a 33-fold increase in the H<sub>2</sub> evolution rate compared to that of bare g-C<sub>3</sub>N<sub>4</sub>.

Despite their strong initial activity for photocatalytic H<sub>2</sub> evolution, metal phosphide co-catalysts suffer from some



drawbacks, which limits their practical use. Under light irradiation in aqueous media, they readily undergo surface oxidation or reconstruction, forming metal oxides or oxyhydroxides and leaching phosphorus, which degrades the true active sites and diminishes long-term performance. Many phosphides also suffer from photocorrosion and poor durability in strongly acidic or alkaline medium. In addition, precise control of stoichiometry and nanostructure is difficult in the case of metal phosphides, which makes them non-uniform, and hence large-scale fabrication of these cocatalysts is challenging. Moreover, phosphorus loss during synthesis or operation raises environmental concerns.

### 3.5. Metal carbides as co-catalysts

Metal carbides have emerged as a fascinating category of cocatalysts effectively used for photocatalytic hydrogen evolution due to their exceptional properties like unique electronic structures, excellent electrical conductivity, and robust chemical stability. These metal carbides exhibit work function values in the range of 4.4–5.0 eV. These properties enable efficient electron transfer to their surfaces, improved separation of photogenerated carriers, and enhanced catalytic activity when coupled with semiconductor photocatalysts.<sup>86,87</sup> In this regard, He *et al.*<sup>88</sup> synthesized Ni<sub>3</sub>C *via* the pyrolysis of nickel acetylacetonate at 250 °C in oleyl amine, and then ground it with g-C<sub>3</sub>N<sub>4</sub> followed by calcination for 1 h at 200 °C to prepare Ni<sub>3</sub>C/g-C<sub>3</sub>N<sub>4</sub> (Fig. 9a). After optimization, the 15 wt% Ni<sub>3</sub>C loaded g-C<sub>3</sub>N<sub>4</sub> showed a H<sub>2</sub> evolution rate of 15.18 μmol h<sup>-1</sup>, which is approximately 116.7 times higher than that of bare g-C<sub>3</sub>N<sub>4</sub>, with a quantum yield of 0.40% (at 420 nm) which is comparable with that of Pt/g-C<sub>3</sub>N<sub>4</sub> (Fig. 9b and c). Furthermore, Tao's group<sup>89</sup> used MAPbI<sub>3</sub> as the semiconductor and Ni<sub>3</sub>C was loaded as a cocatalyst. This approach reported 2362 μmol g<sup>-1</sup> h<sup>-1</sup> H<sub>2</sub> evolution rate, surpassing the rate of bare MAPbI<sub>3</sub> by 55 times (Fig. 9d). The mechanistic pathway of both the catalysts Ni<sub>3</sub>C/g-C<sub>3</sub>N<sub>4</sub> and Ni<sub>3</sub>C/MAPbI<sub>3</sub> is depicted in Fig. 9e and f. In another study, the effect of the Mo<sub>2</sub>C cocatalyst was studied by Liu *et al.*<sup>90</sup> and it was found that Mo<sub>2</sub>C is an outstanding cocatalyst with g-C<sub>3</sub>N<sub>4</sub>. The optimized catalyst 7 wt% 2D/2D Mo<sub>2</sub>C/g-C<sub>3</sub>N<sub>4</sub> exhibited a H<sub>2</sub> production rate of 675.27 μmol g<sup>-1</sup> h<sup>-1</sup>. They compared its activity with 0.5 wt% Pt/g-C<sub>3</sub>N<sub>4</sub> and 7 wt% B/2D Mo<sub>2</sub>C/g-C<sub>3</sub>N<sub>4</sub> and found 3.5- and 5.1-fold higher H<sub>2</sub> evolution rates, respectively. The work suggested the creation of a 2D/2D heterostructure that increases the interfacial contact between the co-catalyst and the semiconductor providing more active sites for electrons to migrate efficiently to the catalyst surface and prevents the recombination of charge carriers (Fig. 9g).

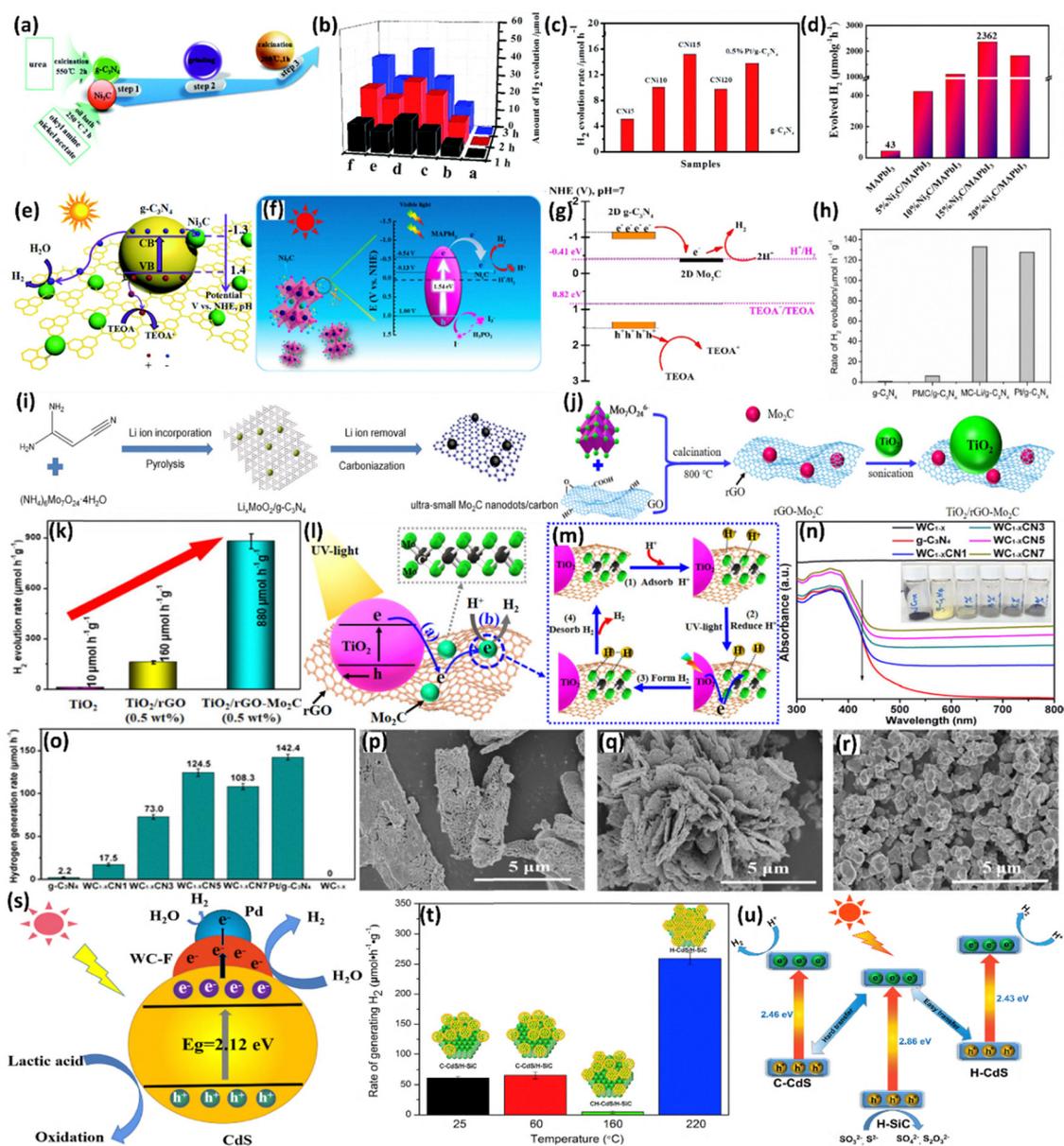
The effect of lithium (Li) incorporation in ultrasmall Mo<sub>2</sub>C nanodots was studied by Yang *et al.*<sup>91</sup> with respect to their efficiency as a cocatalyst for photocatalytic H<sub>2</sub> evolution. It was observed that the H<sub>2</sub> evolution rate was 12.23 times higher than that achieved with pristine MoC loaded on CdS. In addition, the photocatalytic performance of Mo<sub>2</sub>C/CdS was 5.81 times higher than that of Pt/CdS in lactic acid aqueous solution. They also compared the photocatalytic activity with Pt/g-C<sub>3</sub>N<sub>4</sub> and the results are shown in Fig. 9h. Lithium incorporation assisted

synthesis of ultrasmall Mo<sub>2</sub>C nanodots is presented in Fig. 9i. Another study on Mo<sub>2</sub>C was carried out by Liu and coworkers,<sup>90,92</sup> who used the calcination method followed by sonication to synthesize rGO-Mo<sub>2</sub>C/TiO<sub>2</sub> (Fig. 9j). This ternary heterostructure successfully achieved higher photocatalytic activity compared to TiO<sub>2</sub> and rGO/TiO<sub>2</sub> as shown in Fig. 9k. The high photocatalytic activity is ascribed to the synergistic effect arising from rGO nanosheets functioning as electron mediators to enhance photoelectron transfer and Mo<sub>2</sub>C nanoparticles acting as catalytic sites that boost the interfacial hydrogen evolution reaction. In brief, under UV irradiation, electrons from the valence band of TiO<sub>2</sub> are excited to the conduction band, and move *via* rGO sheets, which act as a fast electron promoter, to the Mo<sub>2</sub>C NPs where the H<sup>+</sup> ions get absorbed; the whole mechanism is illustrated in Fig. 9l and m. Another non-noble transition metal compound, tungsten (W) carbide, was studied by Tong and co-workers.<sup>93</sup> They synthesized the tungsten carbide (WC<sub>1-x</sub>) cocatalyst by a calcination process to obtain nanocrystals of size 1.98 ± 0.29 nm followed by their incorporation on g-C<sub>3</sub>N<sub>4</sub> at different weight percentages. As shown in Fig. 9n, UV-Vis absorption spectra showed successful incorporation and a change in color according to the weight percentage, with the color shifting from light to dark with increasing WC<sub>1-x</sub> weight percentage. The photocatalytic H<sub>2</sub> evolution activity of g-C<sub>3</sub>N<sub>4</sub> was significantly improved by loading the WC<sub>1-x</sub> cocatalyst, and the optimized system, WC<sub>1-x</sub>CN5, exhibited the highest H<sub>2</sub> evolution among all weight percentages as shown in Fig. 9o, which can be attributed to the multifunctional role of WC<sub>1-x</sub>, namely, significantly enhancing charge separation and transportation while offering abundant active sites for catalytic reactions.

The effect of different morphologies of WC cocatalysts in photocatalytic H<sub>2</sub> evolution was studied by Dang *et al.*,<sup>94</sup> as shown in Fig. 9p–r. As confirmed from the SEM images, three types of morphologies were designed such as WC-S (sheet), WC-F (flower), and WC-B (bulk), amongst which the 3D flower-like WC exhibited the highest co-catalytic property. Among all three, WC-F/CdS shows the highest H<sub>2</sub> evolution activity of 24 502 μmol h<sup>-1</sup> g<sup>-1</sup>. The most efficient noble metal cocatalyst Pd doped on WC and combined over CdS was compared with the WC-F cocatalyst and it was found that the Pd-doped composite showed a slightly higher activity (1.16 times), as WC-F has a high specific capacitance (12.07 F g<sup>-1</sup>) which helps to trap electrons from the conductance band of CdS. These trapped electrons are consumed for H<sup>+</sup> reduction to H<sub>2</sub> and some unconsumed electrons are transferred to Pd and help in further H<sub>2</sub> evolution as shown in Fig. 9s. Furthermore, Tian *et al.*<sup>95</sup> studied vanadium carbide (VC) integrated on CdS and found that the 15 wt% VC/CdS catalyst showed the highest H<sub>2</sub> photocatalytic activity with a quantum yield of 8.7% (420 nm) which is 12 times higher than that of pure CdS. This approach gives better results than noble metal Ag doped on CdS and almost close results to Pt/CdS. Different wt% VC was integrated over CdS by a sol-gel method.

Research by Peng *et al.*<sup>96</sup> demonstrated how the lattice compatibility between a semiconductor and its cocatalyst can





**Fig. 9** (a) Diagrammatic representation of the  $\text{Ni}_3\text{C}/\text{g-C}_3\text{N}_4$  photocatalyst production process. (b) The photocatalytic  $\text{H}_2$  evolution time courses are as follows: (a)  $\text{g-C}_3\text{N}_4$ , (b) CN15, (c) CN115, (d) CN110, (e) CN120, and (f) 0.5 weight percent Pt/ $\text{g-C}_3\text{N}_4$ . (c) The mean  $\text{H}_2$  evolution rates across  $\text{g-C}_3\text{N}_4$ , 0.5 weight percent Pt/ $\text{g-C}_3\text{N}_4$ , and  $\text{Ni}_3\text{C}/\text{g-C}_3\text{N}_4$  composites. Reproduced with permission.<sup>88</sup> Copyright 2018, Royal Society of Chemistry. (d) Performance of photocatalytic  $\text{H}_2$  evolution in the presence of visible light. Reproduced with permission.<sup>89</sup> Copyright 2019, American Chemical Society. (e) Diagrammatic representation of the  $\text{Ni}_3\text{C}/\text{g-C}_3\text{N}_4$  composite photocatalysts' photo-induced charge separation process. Reproduced with permission.<sup>88</sup> Copyright 2018, Royal Society of Chemistry. (f) Band diagram for the HI-splitting photocatalytic process of  $\text{Ni}_3\text{C}/\text{MAPbI}_3$ . Reproduced with permission.<sup>89</sup> Copyright 2019, American Chemical Society. (g) Schematic representation of electron transfer processes within the 2D/2D  $\text{Mo}_2\text{C}/\text{g-C}_3\text{N}_4$  nanojunction photocatalyst. Reproduced with permission.<sup>90</sup> Copyright 2022, American Chemical Society. (h) The rate of evolution of photocatalytic  $\text{H}_2$  using  $\text{g-C}_3\text{N}_4$ , PMC/ $\text{g-C}_3\text{N}_4$ , MC-Li/ $\text{g-C}_3\text{N}_4$ , and Pt/ $\text{g-C}_3\text{N}_4$ . (i) Synthesis process of  $\text{Mo}_2\text{C}$  nanodots. Reproduced with permission.<sup>91</sup> Copyright 2020, Elsevier. (j) Schematic representation of the rGO- $\text{Mo}_2\text{C}/\text{TiO}_2$  production process. (k)  $\text{H}_2$  evolution rate of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{rGO}$  (0.5 wt%), and  $\text{TiO}_2/\text{rGO-Mo}_2\text{C}$  (0.5 wt%). (l) and (m) Photocatalytic  $\text{H}_2$  evolution mechanism of rGO- $\text{Mo}_2\text{C}/\text{TiO}_2$ . Reproduced with permission.<sup>92</sup> Copyright 2021, American Chemical Society. (n) UV-vis absorption spectra, inset: photograph of different weight percent doped catalyst. (o)  $\text{H}_2$  evolution rate bar graph at different weight percentages. Reproduced with permission.<sup>93</sup> Copyright 2019, American Chemical Society. (p)–(r) SEM image of WC-S (sheet), WC-F (flower), and WC-B (bulk). (s) Mechanism of  $\text{H}_2$  evolution with Pd-WC-F/CdS. Reproduced with permission.<sup>94</sup> Copyright 2021, Elsevier. (t) Comparison of photocatalytic  $\text{H}_2$  evolution rate between different heterointerfaces. (u) Schematic representation of band gap difference between cubic and hexagonal CdS. Reproduced with permission.<sup>96</sup> Copyright 2017, Elsevier.



significantly influence the rate of H<sub>2</sub> evolution. In their study, two different catalysts were synthesized using a hydrothermal method by varying the temperature to form distinct heterointerfaces: one hexagonal-SiC and hexagonal-CdS, and second cubic-CdS and hexagonal-SiC. The lower band gap difference between hexagonal-CdS and hexagonal-SiC facilitates more efficient charge transfer, resulting in a four times increase in the H<sub>2</sub> evolution rate compared to cubic-CdS and hexagonal-SiC (Fig. 9t and u). Overall, this strategy offers a promising route for designing highly efficient photocatalysts through lattice-matched interface engineering.

The drawbacks of metal carbide cocatalysts during the photocatalytic H<sub>2</sub> evolution reaction mainly include their susceptibility to photo-corrosion. In addition, particle aggregation and surface carbon deposition decrease the catalytic activity. Moreover, precise control of stoichiometry, phase purity, and nanoscale morphology is difficult and energy-intensive, making large-scale, low-cost synthesis less practical. These issues like poor long-term durability, interfacial resistance, and scalability constraints remain key hurdles to the reliable use of metal carbide cocatalysts in photocatalytic H<sub>2</sub> evolution.

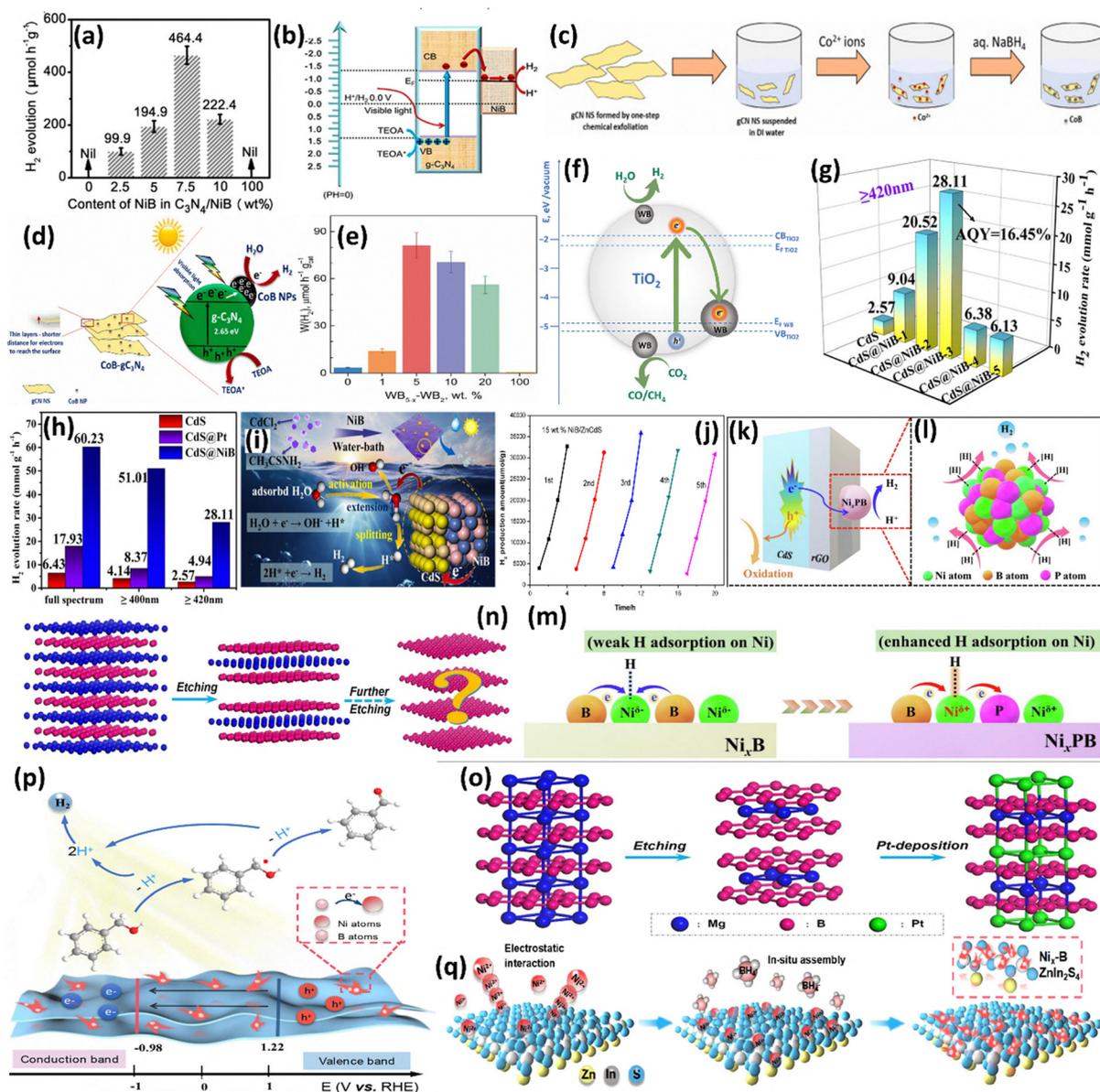
### 3.6. Metal borides as cocatalysts

Noble metals have played a starring role in the dazzling performance of photocatalytic hydrogen evolution reactions, driving innovation with their exceptional efficiency. Yet their exorbitant price and fleeting abundance make them the rare gems of the catalyst world, challenging their widespread adoption in our pursuit of a sustainable future. Platinum, palladium, gold, and silver are noble metals that can enhance the photolytic activity of semiconducting materials. Metal borides with work function values in the range of 4.4–4.6 eV can exhibit similar H<sub>2</sub> evolution performance. Hence, metal borides can be used as a substitute for these metals to improve the photolytic H<sub>2</sub> evolution performance. A study reported<sup>97</sup> the use of nickel boride (NiB), which showed enhanced hydrogen evolution reaction performance, with a hydrogen production rate of 464.4 μmol g<sup>-1</sup> h<sup>-1</sup> and an impressive quantum yield of 10.92% when exposed to 365 nm light. Nickel boride nanoparticles, when used as cocatalysts with the C<sub>3</sub>N<sub>4</sub> photocatalyst and when 7.5 wt% of NiB was combined with C<sub>3</sub>N<sub>4</sub>, led to optimal H<sub>2</sub> evolution (Fig. 10a). The enhancement of H<sub>2</sub> evolution reaction is attributed to the presence of strong B(δ<sup>-</sup>)–Ni(δ<sup>+</sup>)–N(δ<sup>-</sup>) bonds between NiB and C<sub>3</sub>N<sub>4</sub>. The Mott–Schottky test showed that the conduction band potential of C<sub>3</sub>N<sub>4</sub>/NiB7.5 is lower at –1.35 V (Fig. 10b), which means electrons can move from C<sub>3</sub>N<sub>4</sub> to NiB. The photoinduced electrons from C<sub>3</sub>N<sub>4</sub> quickly transfer to NiB through the B(δ<sup>-</sup>)–Ni(δ<sup>+</sup>)–N(δ<sup>-</sup>) bonds. NiB then helps these electrons to reach the active sites for H<sub>2</sub> evolution, while photoinduced holes in C<sub>3</sub>N<sub>4</sub> are consumed by TEOA molecules. This research work showed that metal boride cocatalysts are a good replacement for noble metal cocatalysts. Hence, nickel boride can enhance H<sub>2</sub> evolution. Another report<sup>98</sup> utilizing cobalt demonstrated its potential as an alternative co-catalyst to noble metals. Cobalt boride (CoB) was

loaded onto the semiconductor g-C<sub>3</sub>N<sub>4</sub> nanosheets. The optimized CoB–g-C<sub>3</sub>N<sub>4</sub> composite showcased a significantly higher hydrogen generation rate, approximately 60 times greater than that of the bare g-C<sub>3</sub>N<sub>4</sub> nanosheets. Importantly, it also exhibited commendable stability. To produce gCN nanosheets (NS), melamine and NH<sub>4</sub>Cl were heated together at 550 °C for 4 hours, resulting in exfoliated gCN NS that were then used to load CoB co-catalysts. Initially, 100 mg of gCN NS was mixed with 100 mL of deionized water and sonicated. CoCl<sub>2</sub>·6H<sub>2</sub>O was added to this mixture and stirred for 1 hour to allow Co<sup>2+</sup> ions to adsorb onto the gCN NS. Subsequently, NaBH<sub>4</sub> was added to reduce the Co<sup>2+</sup> ions, forming CoB nanoparticles (NPs) (Fig. 10c). The solution turned dark, indicating successful reduction. After 1 hour, the mixture was centrifuged, washed with deionized water and 2-propanol, and dried under vacuum. Different amounts of Co<sup>2+</sup> (1%, 3%, 5%, 7%, 9%, and 11% by weight) were used to create samples with varying CoB loadings, named 1-CoB-gCN, 3-CoB-gCN, 5-CoB-gCN, 7-CoB-gCN, 9-CoB-gCN, and 11-CoB-gCN. At 9 wt%, the hydrogen evolution reaction is maximum, and beyond 11 wt%, the co-catalyst covers the surface of the gCN nanosheet which causes shielding and affects the absorption of photons; beyond 5 wt% loading there is increment in the H<sub>2</sub> evolution. The plausible mechanism of H<sub>2</sub> evolution is shown in Fig. 10d. A report by Kurenkova *et al.*<sup>99</sup> showed metal borides to be effective cocatalysts for oxide semiconductors. This research shows that tungsten boride (WB) increases the hydrogen evolution reaction. Loading the WB co-catalyst on TiO<sub>2</sub> semiconductors increases the rate of H<sub>2</sub> evolution by 23 times in comparison to pure TiO<sub>2</sub>. The H<sub>2</sub> evolution reaction rate increases as the concentration of the co-catalyst increases from 0 wt% to 5 wt% and the maximum activity recorded is 81 μmol h<sup>-1</sup> g<sub>cat</sub><sup>-1</sup> (Fig. 10e). When TiO<sub>2</sub> is exposed to light, electrons move from the valence band to the conduction band, creating electron–hole pairs. In pure TiO<sub>2</sub>, these charge carriers recombine rapidly, resulting in a short lifespan for the electrons and holes. By adding WB<sub>5-x</sub> particles to TiO<sub>2</sub>, these particles capture some of the electrons from the TiO<sub>2</sub> conduction band (Fig. 10f). This transfer of electrons to the WB<sub>5-x</sub> particles helps to separate the electron–hole pairs more effectively. As the amount of WB<sub>5-x</sub> particles increases, more contact sites between TiO<sub>2</sub> and WB<sub>5-x</sub> are created, enhancing the efficiency by decreasing the gap between the conducting band and the valence band. Boride creates a Fermi level. Tungsten boride (WB) increases the lifetime of photogenerated charge carriers and photocatalytic activity.

A drawback of loading co-catalysts on semiconductors is the shielding of the active site of the semiconductor (TiO<sub>2</sub>). CdS@NiB composites (0.005, 0.01, 0.015, 0.02, 0.03) were prepared by Lv *et al.*<sup>100</sup> by the water bath method; the harsh conditions of hydrothermal synthesis reduce CdS activity, while a one-step water bath method preserves the active sites of CdS. The rate of H<sub>2</sub> evolution was reported as 28 112 μmol h<sup>-1</sup> g<sup>-1</sup> with an improvement of 5.7 times for CdS@Pt and 11 times for pure CdS (Fig. 10g and h). The enhanced performance of H<sub>2</sub> evolution supported by ‘density functional theory’ calculations





**Fig. 10** (a)  $H_2$  evolution rate of NiB/ $C_3N_4$  at different wt%. (b) Mechanism of  $H_2$  evolution with g- $C_3N_4$ /NiB7.5. Reproduced with permission.<sup>97</sup> Copyright 2018, American Chemical Society. (c) Schematic representation of the synthesis and formation of the CoB-g- $C_3N_4$  composite. (d) Schematic illustration of the photocatalytic  $H_2$  generation mechanism of CoB-g- $C_3N_4$  composites. Reproduced with permission.<sup>98</sup> Copyright 2024, Elsevier. (e)  $H_2$  evolution rate at different wt% of  $WB_{5-x}-WB_2TiO_2$ . (f) Mechanism of electron transfer and  $H_2$  evolution on  $WB_{5-x}/TiO_2$ . Reproduced with permission.<sup>99</sup> Copyright 2024, Elsevier. (g) and (h)  $H_2$  evolution rate at different wt% of CdS@NiB and with pure CdS and CdS@Pt. (i) Schematic representation of the plausible mechanism of CdS@NiB  $H_2$  evolution. Reproduced with permission.<sup>100</sup> Copyright 2022, American Chemical Society. (j) Recyclability  $H_2$  evolution rate of 15 wt% NiB/ZnCdS over five cycles. Reproduced with permission.<sup>101</sup> Copyright 2022, Elsevier. (k) and (l) Mechanism of  $H_2$  evolution on  $Ni_xPB-rGO/CdS$ . (m) Schematic representation of electron transport on  $Ni_xB$  and  $Ni_xPB$ . Reproduced with permission.<sup>102</sup> Copyright 2022, Elsevier. (n) Synthesis process of layered boron. (o) Synthesis process of exfoliated Pt doped  $MgB_2$ . Reproduced with permission.<sup>103</sup> Copyright 2021, American Chemical Society. (p)  $H_2$  evolution mechanism over  $Ni_x-B/ZnIn_2S_4$ . (q) Diagrammatical representation of the synthesis scheme of  $Ni_x-B/ZnIn_2S_4$ . Reproduced with permission.<sup>104</sup> Copyright 2022, American Chemical Society.

reveals that incorporating NiB into CdS results in a strong electron coupling effect that enhances electron participation in  $H_2$  production. The d-band center of Ni atoms in the CdS@NiB composite shifts from  $-1.96$  eV to  $-1.84$  eV upon adsorption of water. This shift promotes the movement of electrons across the Fermi level, improving the electron transfer from NiB  $\rightarrow$  CdS  $\rightarrow$   $H_2O$ . This interaction results in a significant increase in

the bond length of  $H_2O$  from  $0.975$  Å to  $0.986$  Å and an increase in the H-O-H bond angle from  $104.274^\circ$  to  $105.365^\circ$ . This facilitates decomposition into hydrogen and hydroxyl ions. Also, the incorporation of NiB narrows the band gap of CdS from  $2.12$  eV to  $1.91$  eV. The charge density redistribution in CdS@NiB- $H_2O$  leads to substantial charge accumulation around CdS and  $H_2O$ . Water molecules preferentially adsorb



on the Cd atoms of CdS. The electron-rich sites on CdS facilitate the transfer of hydrogen atoms ( $H^*$ ) and hydroxyl ions ( $OH^-$ ), resulting in the formation of adsorbed hydrogen intermediates ( $H_2O + e^- \rightarrow H^* + OH^-$ ). These intermediates then combine to form  $H_2$  ( $2H^* + 2e^- \rightarrow H_2$ ), leading to the release of a substantial amount of  $H_2$  from the CdS@NiB composite (Fig. 10i). Despite the efficient performance of CdS in photocatalytic hydrogen production, its practical application is hindered due to its photoinduced corrosive nature. Song *et al.*<sup>101</sup> reported the synthesis of ZnCdS by a hydrothermal method. Since the rapid recombination of photogenerated electrons and holes limits the photocatalytic efficiency of ZnCdS, the cocatalyst NiB at different wt% was doped into it by calcination method at 773 K. The highest hydrogen evolution was reported at 15 wt%, being 17 times higher than that achieved with pure ZnCdS with a rate of  $8137.1 \mu\text{mol h}^{-1} \text{g}^{-1}$ ; a further increase in wt% led a decrease in  $H_2$  evolution. The recycling stability test over five cycles showed that hydrogen production remained high and relatively stable throughout all cycles (Fig. 10j).

Furthermore, Long *et al.*<sup>102</sup> reported another approach to improve the  $H_2$  evolution, by increasing the efficiency of the cocatalyst. Long *et al.* incorporated P in between  $Ni_xB$ , loaded it on reduced graphene oxide, and combined it with the CdS  $Ni_xPB-rGO/CdS$  photocatalyst. This improvement is attributed to the decreased electron density of the Ni atoms in  $Ni_xB$ , enhancing the hydrogen-adsorption capacity at the Ni active sites and thereby boosting the hydrogen-evolution process by 1.8 times compared to  $Ni_xB-rGO/CdS$  and by 9.9 times compared to  $rGO/CdS$ . Upon exposure to visible light, the  $Ni_xPB-rGO/CdS$  photocatalyst generates electron-hole pairs within the CdS component (Fig. 10k and l). The electrons and holes are then separated and migrate to the Ni active sites on the  $Ni_xPB$  cocatalysts. Hydrogen is produced through this process, with the rGO nanosheets facilitating the efficient transfer of electrons to the Ni sites (Fig. 10m). The  $Ni_xPB-rGO/CdS$  photocatalysts were synthesized using a simple chemical reduction method at room temperature. This process involved mixing cadmium sulfide with graphene oxide using sonication, followed by the *in situ* formation of amorphous  $Ni_xPB$ . The mass ratio of the  $Ni_xPB$  cocatalyst to CdS was set at 5 wt%. The molar ratios of phosphorus (P) to boron (B) were varied at 0.2, 0.5, 1, 2, and 5.

Besides, a novel material  $MgB_4$  MXene has been developed to drive the  $H_2$  evolution as well. Xiao's group<sup>103</sup> found that doping of Pt on nanosheets (Pt- $MgB_4$ ) results in an enhanced ability to absorb and utilize solar energy due to electron trapping by Pt and the abundant terminal metal sites on their surface and exceptional metallic-layer conductivity.  $MgB_4$  MXene-like nanosheets are achieved by exfoliation and selective etching. The  $MgB_2$  powder is first dispersed and exfoliated using ultrasound and then subjected to selective etching with HCl. HCl is used to remove Mg atoms (Fig. 10n and o). The recyclability test over four cycles showed no decreases in Pt deposit, and  $H_2$  evolution in the 1st cycle was  $1250 \mu\text{mol}$  and  $1175 \mu\text{mol}$  in the 4th cycle. Without platinum (Pt), the  $MgB_4$

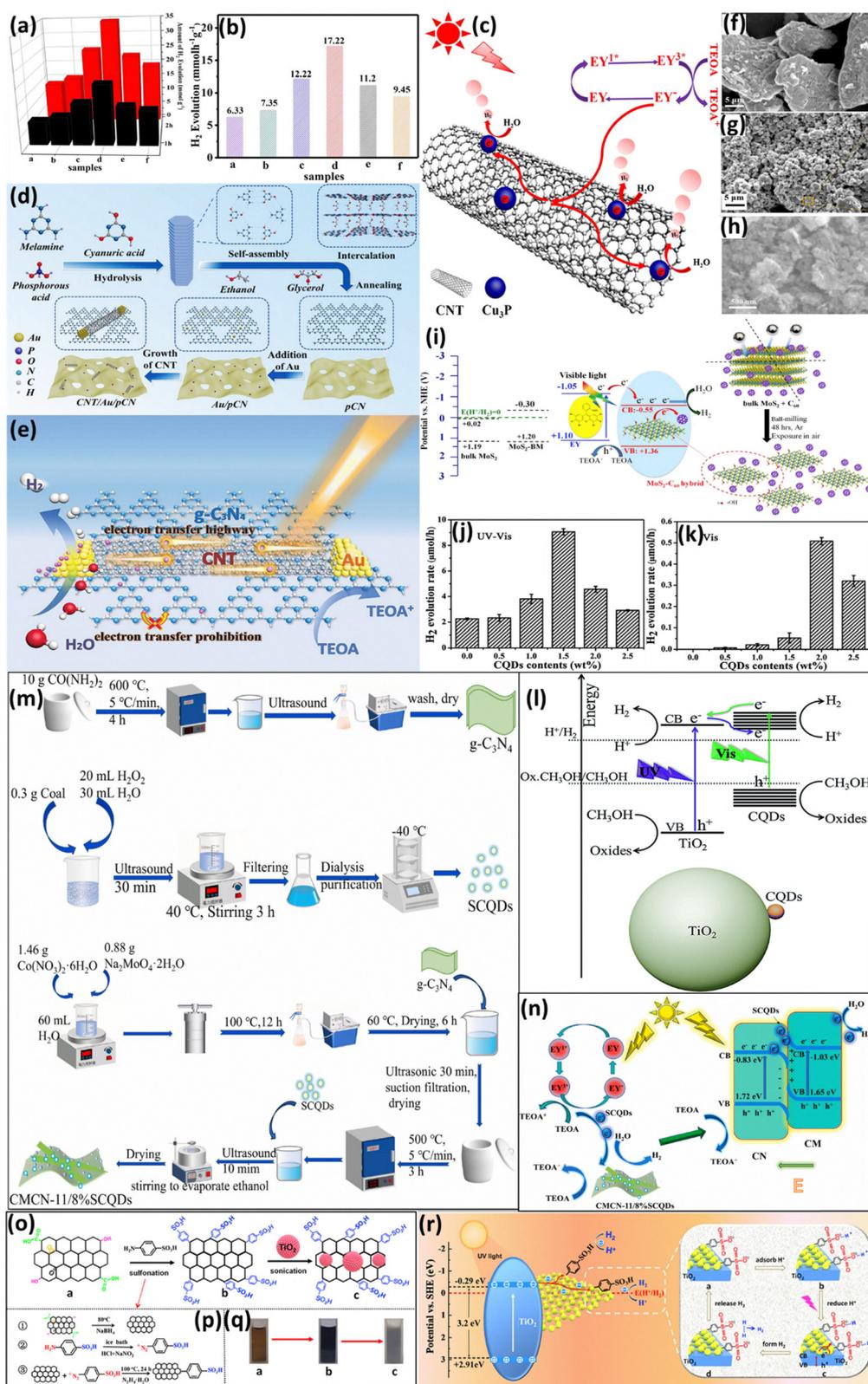
MXene-like nanosheets generate about  $120 \mu\text{mol}$  of hydrogen after 4 hours, which is roughly 10% of the amount produced by the Pt-deposited  $MgB_4$  nanosheets. The photocatalytic activity of these Pt-deposited  $MgB_4$  nanosheets is comparable to that of Pt-deposited P25 under ultraviolet light. Additionally,  $MgB_4$  MXene-like nanosheets with Pt exhibit superior performance under visible light because their smaller bandgap allows for better solar energy absorption and utilization. The valence band maximum shifts upward in Pt- $MgB_4$  due to the influence of Pt atoms, which introduces surface defects or oxygen vacancies. This shift enhances visible light absorption and improves photocatalytic hydrogen production. Additionally, UV-Vis DRS results indicate that Pt- $MgB_4$  has a characteristic bandgap, but absorption in the infrared region suggests an even smaller bandgap. The top of the valence band (VB) is approximately 0.34 eV for  $MgB_4$  and around 1.52 eV for Pt-deposited  $MgB_4$ .

Water splitting is the most common way for  $H_2$  evolution; Li *et al.*<sup>104</sup> reported organic transformation for  $H_2$  by using a  $Ni_x-B$  cocatalyst doped on 2D ultrathin  $ZnIn_2S_4$  nanosheets. The rate of  $H_2$  production reported with a quantum yield of 24% is 3.1 times higher than that of the traditional  $H_2$  photocatalytic half-reaction. When  $ZnIn_2S_4$  absorbs light energy, electrons are promoted from the valence band to the conduction band, generating electron-hole pairs. Aromatic alcohols absorbed on the surface capture holes and the holes oxidize C-H bonds to form radicals and protons, and these radicals are converted into aldehydes with the concomitant release of protons. The electrons in the conduction band are transferred to the  $Ni_x-B$  cocatalyst. Protons are reduced by the cocatalyst into  $H_2$  gas. The nanosheet facilitates quick separation of electron-hole pairs and cocatalyst reduction (Fig. 10p). The nanosheets are prepared by a facile reflux method and the  $ZnIn_2S_4/Ni_x-B$  heterostructure with different compositions ( $x = 0.3, 0.5, 1, 1.5\%$ ) is prepared by an *in situ* growth method as shown in Fig. 10q. The  $ZnIn_2S_4$  nanosheet has a negative zeta potential leading to the adsorption of nickel ions due to electrostatic attraction. Sodium borohydride reduces nickel ions into nickel-boron compounds like  $Ni_x-B$ . For comparison,  $ZnIn_2S_4/Pt$ ,  $ZnIn_2S_4/NiS$ , and  $ZnIn_2S_4/Ni(OH)_2$  were prepared utilizing photodeposition, reflux and coprecipitation method, respectively. The photoactivity performance test shows that the performance of  $ZnIn_2S_4/Ni_x-B$  0.5% is higher than that of  $ZnIn_2S_4/NiS$ ,  $ZnIn_2S_4/Ni(OH)_2$  and  $ZnIn_2S_4/Pt$ . High-temperature synthesis to control stoichiometry and phase variations with very small stoichiometric differences hinders their practical application as a cocatalyst.

### 3.7. Carbon-based cocatalysts

Carbon-based materials, including carbon nanotubes (CNTs), activated carbon, and graphene, are highly abundant on Earth. Carbon-based cocatalysts possess photocatalytic properties similar to those of noble metals, enhanced by features such as extended light absorption that provides a localized photo-thermal effect and a large surface area. They show a work function of 4.5–4.7 eV which is favorable for electron uptake from the CB of the semiconductor for the  $H_2$  evolution reaction.





**Fig. 11** (a) H<sub>2</sub> evolution study by time. (b) H<sub>2</sub> evolution study of different systems (a) Cu<sub>3</sub>P, (b) Cu<sub>3</sub>P-0.5% CNT, (c) Cu<sub>3</sub>P-1% CNT, (d) Cu<sub>3</sub>P-3% CNT, (e) Cu<sub>3</sub>P-5% CNT, (f) Cu<sub>3</sub>P-10% CNT. (c) Mechanism of H<sub>2</sub> evolution via Cu<sub>3</sub>P-CNT. Reproduced with permission.<sup>105</sup> Copyright 2018, American Chemical Society. (d) Synthesis scheme of CNT/Au/g-C<sub>3</sub>N<sub>4</sub>. (e) Mechanism of H<sub>2</sub> evolution via CNT/Au/g-C<sub>3</sub>N<sub>4</sub>. Reproduced with permission.<sup>106</sup> Copyright 2022, John Wiley and Sons. (f)–(h) SEM images of bulk MoS<sub>2</sub> and ball milled MoS<sub>2</sub>-C<sub>60</sub>-BM3. (i) The left side shows the mechanism of H<sub>2</sub> evolution via the MoS<sub>2</sub>-C<sub>60</sub> hybrid and the right side shows the ball milling process. Reproduced with permission.<sup>107</sup> Copyright 2018, Elsevier. (j) and (k) H<sub>2</sub> generation rate study via UV-Vis and visible light of pure TiO<sub>2</sub> and different wt% loaded CQDs. (l) Plausible mechanism of photocatalytic H<sub>2</sub> evolution via CQDs/TiO<sub>2</sub>.



Reproduced with permission.<sup>108</sup> Copyright 2015, American Chemical Society. (m) Schematic representation of the synthesis scheme of CMCN-11/8%SCQDs. (n) Mechanism of photocatalytic H<sub>2</sub> evolution of CMCN-11/8%SCQDs. Reproduced with permission.<sup>109</sup> Copyright 2024, Elsevier. (o) Schematic representation of the synthesis scheme of rGO-SO<sub>3</sub>H/TiO<sub>2</sub>. (p) Formation scheme of rGO-SO<sub>3</sub>H. (q) Digital pictures of (a) rGO, (b) rGO-SO<sub>3</sub>H, and (c) rGO-SO<sub>3</sub>H/TiO<sub>2</sub>. (r) Mechanism of photocatalytic H<sub>2</sub> evolution *via* rGO-SO<sub>3</sub>H/TiO<sub>2</sub>. Reproduced with permission.<sup>110</sup> Copyright 2022, Elsevier.

Additionally, carbon materials are inert and thermally stable. They serve as nucleation sites for semiconductor nanoparticle growth due to their defect-rich surfaces that facilitate nucleation. Moreover, they increase the number of active sites for photocatalytic activity. Typically, reactions occur at the interface of the semiconductor material and the cocatalyst, but with carbon materials, reactions also take place on the surface of the carbon surrounding the interface. This section discusses four carbon-based cocatalysts: carbon nanotubes (CNTs), carbon quantum dots, graphene, and fullerene (C<sub>60</sub>).

In this regard, Shen *et al.*<sup>105</sup> reported a semiconductor-free Cu<sub>3</sub>P-supported CNT system and the Cu<sub>3</sub>P-CNT was synthesized *via* a sol-gel method followed by thermal reduction. They reported a quantum yield of 10.23% with Cu<sub>3</sub>P-CNT-EY (Eosin Y). This evolution rate was 2.5 times more than that of bare Cu<sub>3</sub>P and higher than that of some other reported systems like Pt/RGO/TPPH, [ZnTm-PyP]<sup>4+</sup>-MoS<sub>2</sub>/RGO and ZnTCPP-MoS<sub>2</sub>/TiO<sub>2</sub>. The photocatalytic reaction setup used 50 mg of catalyst Cu<sub>3</sub>P-CNT and 18 mg of EY in a 15% volume solution of TEOA. Different wt% of CNTs was loaded on Cu<sub>3</sub>P, and the activity increased until 3 wt% and also showed the highest H<sub>2</sub> evolution rate as represented in Fig. 11a and b. Fig. 11a shows a time study of H<sub>2</sub> evolution; from the starting hours Cu<sub>3</sub>P-3%CNT shows the highest yield. A plausible mechanism of this system is shown in Fig. 11c, in which first in the presence of light, the EY dye electron got excited to a singlet state, underwent intersystem crossing and then was transferred to the triplet state. Then in the presence of TEOA, it got reduced to EY<sup>-</sup>. The non-covalent interaction between CNT and EY dye facilitated the electron transfer from EY<sup>-</sup> to Cu<sub>3</sub>P. This further transfers electrons for H<sub>2</sub>O reduction to H<sub>2</sub>.

In another study, Xia *et al.*<sup>106</sup> reported that CNTs can enhance the H<sub>2</sub> evolution rate of the noble metal incorporated catalyst, Au/g-C<sub>3</sub>N<sub>4</sub>. Fig. 11d shows the synthesis scheme of the CNT/Au/g-C<sub>3</sub>N<sub>4</sub> catalyst. First, g-C<sub>3</sub>N<sub>4</sub> was synthesized *via* hydrolysis followed by annealing and then Au nanoparticles were incorporated on g-C<sub>3</sub>N<sub>4</sub> sheets *via* a deposition-precipitation method. Thereafter, the CNT was grown on Au/g-C<sub>3</sub>N<sub>4</sub> *via* a chemical vapor deposition method. The use of CNTs for the modification of the catalyst helps as an electron highway as shown in Fig. 11e and as electrons jump from the valence band to the conductance band when g-C<sub>3</sub>N<sub>4</sub> absorbs light, these electrons diffuse on Au *via* CNTs and the gathered electrons reduce water to H<sub>2</sub> gas with an evolution rate of 950 μmol g<sup>-1</sup> h<sup>-1</sup> which is 3-fold higher than that of the Au/g-C<sub>3</sub>N<sub>4</sub> binary counterpart.

Another approach to increase photocatalytic activity *via* carbon based co-catalysts is using C<sub>60</sub>. Guan *et al.*<sup>107</sup> reported a C<sub>60</sub>/MoS<sub>2</sub> photocatalyst which showed an enhanced H<sub>2</sub> evolution in the presence of EY. By using the ball milling technique,

C<sub>60</sub> was deposited on the edge of MoS<sub>2</sub>. The ball milling technique helped to convert bulk MoS<sub>2</sub> to small-size sheets as represented in SEM images (Fig. 11f-h), which show that before ball milling the size of MoS<sub>2</sub> was 10 micrometers and after ball milling, the size was reduced to 1 micrometer. This phenomenon is also diagrammatically shown in Fig. 11i, which shows that in bulk form MoS<sub>2</sub> its edges were captured by C<sub>60</sub> while after ball-milling MoS<sub>2</sub> breaks into thin sheets; the mechanism of the photocatalytic H<sub>2</sub> evolution of the MoS<sub>2</sub>/C<sub>60</sub>-EY catalyst is also shown in Fig. 11i. It also followed the same pathway as discussed in the above cases; the EY dye first transfers electrons to MoS<sub>2</sub> and then to C<sub>60</sub> which is present on the edges, which further helps in H<sub>2</sub> evolution from H<sub>2</sub>O. The authors tried many morphologies for the analysis: bulk MoS<sub>2</sub> + EY showed a zero H<sub>2</sub> evolution rate and the ball milled MoS<sub>2</sub> showed H<sub>2</sub> evolution, with MoS<sub>2</sub>-(2.8 wt%) C<sub>60</sub>-BM + EY showing the highest H<sub>2</sub> evolution of all tried morphologies. The ball-milled MoS<sub>2</sub>-(2.8 wt%) C<sub>60</sub> exhibited a higher H<sub>2</sub> evolution rate than the physically blended MoS<sub>2</sub>/C<sub>60</sub>. Lian *et al.*<sup>108</sup> reported C<sub>60</sub>/CdS/TiO<sub>2</sub> which was synthesized *via* evaporation-induced self-assembly (EISA). Different wt% of C<sub>60</sub> (0.25–1.5 wt%) was integrated out of which 0.50 wt% showed the highest H<sub>2</sub> evolution rate of 6.03 μmol h<sup>-1</sup>, with a quantum yield of 2% at 420 nm; however, without C<sub>60</sub> integration, H<sub>2</sub> evolution was quite low, *i.e.*, 0.71 μmol h<sup>-1</sup>. Hence, C<sub>60</sub> acted as a transit station when electrons get excited from the valence band to the conduction band in the presence of visible light from both TiO<sub>2</sub> and CdS; C<sub>60</sub> acted as an electron acceptor from both CdS and TiO<sub>2</sub> and used these trapped electrons for the H<sub>2</sub> evolution.

Another type of carbon-based co-catalyst is carbon quantum dots (CQDs). Yu *et al.*<sup>108</sup> synthesized CQDs/TiO<sub>2</sub> *via* a hydrothermal method which showed a 4-fold higher H<sub>2</sub> evolution rate compared to that of bare TiO<sub>2</sub>. Different wt% of CQDs (*i.e.*, 0.5 to 2.5 wt%) was integrated on TiO<sub>2</sub>, and out of these 1.5 wt% integration showed the highest photocatalytic activity. However, a further increase in integration wt% led to a decrease in the H<sub>2</sub> evolution rate due to a decrease in active sites as doping increased. Photocatalytic experiments were carried out under UV-Vis and visible light as shown in Fig. 11j and k. Also, UV-vis doping with 1.5 wt% and visible light doping with 2 wt% showed the highest photocatalytic activity. In this approach, CQDs act as photosensitizers and also help to slowdown the charge recombination rate by accepting electrons from the TiO<sub>2</sub> conduction band (Fig. 11l). Xu *et al.*<sup>109</sup> synthesized sulfur carbon quantum dots (SCQDs) and integrated them on CoMoO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> (CMCN). The detailed synthesis scheme is shown in Fig. 11m, in which different wt% of SCQDs was loaded on the 1:1 mass ratio of CM/CN, and the CMCN-11/8%SCQDs exhibited the highest H<sub>2</sub> evolution rate of 4916.63 μmol g<sup>-1</sup> h<sup>-1</sup> at 9 pH with a quantum yield of 1.78%. This yield was possible due to the fast electron



transfer provided by the EY dye while absorbing light which then transferred to SCQDs that slowed the recombination rate. CM/CN also absorbs light and forms an n-type semiconductor and this is converted into an S-type heterojunction *via* bending of Fermi levels. This alignment creates an internal electric field which helps move electrons in the right direction. Hence, electrons from SCQDs and CN transfer to CM and these electrons react with H<sub>2</sub>O molecules to produce H<sub>2</sub> gas (Fig. 10n).

The last type of carbon-based cocatalyst is graphene. Wang *et al.*<sup>110</sup> reported TiO<sub>2</sub> doped with edge sulfonated graphene (rGO-SO<sub>3</sub>H). The rGO-SO<sub>3</sub>H catalyst was prepared by reduction of graphene in the presence of NaBH<sub>4</sub>, followed by acid sulfonation at the graphene edges *via* a diazotization reaction. This process involved the formation of a diazonium salt and subsequent sonication to modify TiO<sub>2</sub>, resulting in the formation of rGO-SO<sub>3</sub>H/TiO<sub>2</sub> (Fig. 10o–q). Under UV light, electrons are excited to the conduction band of TiO<sub>2</sub> and then transferred to graphene and then to the -SO<sub>3</sub>H group on benzene which forms the -SO<sub>3</sub><sup>-</sup> ion which first absorbs H<sup>+</sup> and then reduces it by donating electrons. As shown in the last step of Fig. 10r, H<sub>2</sub> gas is released through the combination of two hydrogen atoms leaving the -SO<sub>3</sub><sup>-</sup> ion available to repeat the reaction. A H<sub>2</sub> evolution rate of 197.1 μmol h<sup>-1</sup> g<sup>-1</sup> was reported.

Regardless of the abundant surface area for photocatalytic H<sub>2</sub> evolution provided by the carbon based cocatalysts, they show limited intrinsic catalytic activity, often requiring additional modifications like heteroatom doping, hybridization with metals, defect creation, and modulated dimensions to achieve competitive activity.

## 4. Summary and future perspectives

In summary, the photocatalytic H<sub>2</sub> evolution efficiency of different semiconductors may be enhanced by the use of cocatalysts, which is promising and extremely effective and has gathered significant attention in recent decades. The practical use of noble-metal-based cocatalysts in large-scale energy applications is hampered by their high cost and restricted availability, despite their exceptional performance. Thus, significant research efforts have gone into creating effective cocatalysts devoid of noble metals. Different non-noble metal based cocatalysts have drawn a lot of interest among the alternatives because of their great charge carrier-capturing capacity, low cost, wide availability, and outstanding stability, which makes them appealing options for photocatalytic applications. This article provides an in-depth overview of the key developments in widespread and new generation cocatalysts for boosting the H<sub>2</sub> evolution performance by photocatalytic water splitting, including the effect of cocatalysts in enhancing the photocatalytic H<sub>2</sub> evolution reaction, their classification as metal nanoparticles, single atom cocatalysts, bimetallic alloy cocatalysts, metal oxide cocatalysts, transition metal dichalcogenide cocatalysts, metal phosphide cocatalysts, metal carbide cocatalysts, carbon based cocatalysts, and metal

boride cocatalysts, and their modification over various well-known photocatalysts.

Despite tremendous advancements in the development of stable and effective cocatalysts and in the understanding of their underlying mechanisms, considerable challenges remain in substantially enhancing the efficiency and ensuring long-term stability of photocatalytic water splitting systems using these cocatalysts. The following problems need to be addressed.

(i) Since the cocatalysts are essential for attaining effective photocatalytic H<sub>2</sub> evolution with semiconductors, their performance needs to be greatly improved in order to further improve the efficiency and durability of the cocatalysts. Continuous research efforts are therefore essential in identifying and designing high-performance cocatalysts. Among the strategies to be emphasized, one is improving the intrinsic properties of existing cocatalysts, such as enhancing their charge-trapping ability, increasing the number of surface active sites, and reducing the overpotentials of surface reactions. These goals can be addressed through appropriate modification attempts. Another one is discovering innovative forms of cocatalysts that can offer superior charge extraction and can optimize adsorption and dissociation of water molecules.

(ii) Though many efforts have been devoted to lab-scale H<sub>2</sub> evolution using co-catalyst–semiconductor systems, their scalability for industrial-scale H<sub>2</sub> production remains a formidable challenge. The practical implementation of these catalytic systems at the scale required for significant H<sub>2</sub> production needs an understanding of materials, engineering, and economic bottlenecks. However, the uniform and reproducible deposition of the cocatalyst over a large area of the semiconductor is necessary; state-of-the-art techniques like photodeposition, electrodeposition, and atomic layer deposition (ALD) can be adopted to control particle size and distribution. In addition, the design and architecture of multicomponent systems like type-II, Z-scheme, and S-scheme heterojunctions can be fabricated in which selective deposition of cocatalysts over the surface of the collective component will lead to improved photocatalytic H<sub>2</sub> production. In this regard, a large-scale prototype of a photocatalytic water-splitting device has indeed been demonstrated by Nishiyama *et al.*, who reported safe, continuous hydrogen production on a 100 m<sup>2</sup> outdoor photocatalytic panel reactor array, operating for several months under natural sunlight and achieving a solar-to-hydrogen efficiency of up to 0.76% while autonomously recovering hydrogen using a commercial polyimide membrane.<sup>111</sup> Although this represents the first field-scale implementation of photocatalytic overall water splitting, the technology remains at the research stage. For a commercial scale, efficiencies must reach 5–10%, photocatalyst lifetime and visible-light activity must improve, and cost-effective reactor and gas-separation designs are still needed before commercialization. Ongoing global efforts are addressing these challenges to translate such demonstrations into economically viable industrial systems.

(iii) Comprehensive research is necessary to fully understand the photocatalytic process pertaining to the migration



pathway of photogenerated charge carriers and the photocatalytic reaction active sites. There is currently no concrete proof of the migratory route and real active locations of photogenerated electrons and holes. Additionally, the literature provides conflicting pathways for interfacial charge transfer. For instance, the photocatalytic H<sub>2</sub>-production activity may be increased by intimate contact or spatial charge separation by reduction cocatalysts. These problems should be carefully examined since they are important for the choice of components and the design of the photocatalytic system.

(iv) Much of the existing knowledge is based on conjectural interpretations, and the fundamental processes underpinning the enhanced performance of the photocatalytic hydrogen evolution reaction are still controversial. Therefore, in-depth mechanistic studies are crucial. By allowing for real-time monitoring of charge transfer at the cocatalyst/semiconductor interface and surface catalytic processes, operando techniques like *in situ* X-ray photoelectron spectroscopy (XPS), *in situ* transmission electron microscopy (TEM), and *in situ* extended X-ray absorption fine structure (EXAFS) provide insightful information. Furthermore, theoretical computations are essential for directing the choice of the best cocatalysts and clarifying the processes behind the noted increase in photocatalytic H<sub>2</sub> evolution activity after cocatalyst deposition.

## Conflicts of interest

The authors declare no conflict of interest.

## Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this review.

## Acknowledgements

The authors are thankful to IIT Mandi for the facilities provided. We are thankful to Dr Ashish Kumar, Sardar Patel University Mandi, Himachal Pradesh, India, for his insightful inputs while compiling this article. BPM acknowledges the Science and Engineering Research Board (SERB), Department of Science and Technology (DST), Government of India, for the National Postdoctoral Fellowship (NPDF). VK acknowledges the DST-SERB project (CRG/2022/003559) for financial support.

## References

- S. Nishioka, F. E. Osterloh, X. Wang, T. E. Mallouk and K. Maeda, *Nat. Rev. Methods Primers*, 2023, **3**, 42.
- Q. Wang and K. Domen, *Chem. Rev.*, 2020, **120**, 919–985.
- D. Sharma, P. Choudhary, S. Kumar and V. Krishnan, *Small*, 2023, **19**, 2207053.
- A. Kumar and V. Krishnan, *Adv. Funct. Mater.*, 2021, **31**, 2009807.
- B. P. Mishra and K. Parida, *J. Mater. Chem. A*, 2021, **9**, 10039–10080.
- M. Yue, H. Lambert, E. Pahon, R. Roche, S. Jemei and D. Hissel, *Renewable Sustainable Energy Rev.*, 2021, **146**, 111180.
- X. Yi, T. Lu, Y. Li, Q. Ai and R. Hao, *Renewable Sustainable Energy Rev.*, 2025, **210**, 115147.
- J. Yang, X. Zheng, S. S. A. Shah, C. Wang, X. Li, Z. Yan and L. Peng, *Carbon Energy*, 2025, e695.
- H. Karibe, S. Sair, A. Faik and H. Ait Ousaleh, *Int. J. Hydrogen Energy*, 2025, **133**, 200–213.
- M. A. Bashir, J. Tuo, J. Weidman, Y. Soong, M. L. Gray, F. Shi and P. Wang, *Energy Adv.*, 2025, **4**, 330–363.
- H. Wang, E. Harkou, A. Constantinou, S. M. Al-Salemc, G. Manos and J. Tang, *Chem. Soc. Rev.*, 2025, **54**, 2188–2207.
- P. Saini, D. Jampaiah, S. Periasamy, A. P. Kulkarni and S. K. Bhargava, *Chem. Commun.*, 2025, **61**, 6027–6054.
- G. Zou, Q. Wang, G. Ye, Z. Pan, S. Wang, M. Anpo and G. Zhang, *Adv. Funct. Mater.*, 2025, **35**, 2420899.
- H. Zhuzhang, X. Liang, J. Li, S. Xue, Y. Lin, B. Sa, S. Wang, G. Zhang, Z. Yu and X. Wang, *Angew. Chem., Int. Ed.*, 2025, **64**, e202421861.
- Q. Wang, D. Zheng, Z. Pan, W. Xing, S. Wang, Y. Hou, M. Anpo and G. Zhang, *Adv. Funct. Mater.*, 2025, 2501889.
- Y. Zou, S. Li, D. Zheng, J. Feng, S. Wang, Y. Hou and G. Zhang, *Sci. China Chem.*, 2024, **67**, 2215–2223.
- J. Yang, S. Tian, Z. Song, Y. Hao and M. Lu, *Coord. Chem. Rev.*, 2025, **523**, 216257.
- S. Raza, I. Sadiq, S. Shaheen, M. Saniya and T. Ahmad, *Sustainability Sci. Technol.*, 2025, **2**, 012002.
- S. A. Ali, S. Majumdar, P. K. Chowdhury, S. M. Alshehri and T. Ahmad, *ACS Appl. Energy Mater.*, 2024, **7**, 7325–7337.
- A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- A. Kumar, A. Kumar and V. Krishnan, *ACS Catal.*, 2020, **10**, 10253–10315.
- A. Kumar, P. Choudhary, A. Kumar, P. H. C. Camargo and V. Krishnan, *Small*, 2022, **18**, 2101638.
- M. A. Khan, S. Mutahir, I. Shaheen, Y. Qunhui, M. Bououdina and M. Humayun, *Coord. Chem. Rev.*, 2025, **522**, 216227.
- B. Guan, J. Chen, Z. Li, Z. Zhuang, Y. Chen, Z. Ma, J. Guo, C. Zhu, X. Hu, S. Zhao, H. Dang, L. Chen, K. Shu, Z. Guo, K. Shi, Y. Li, C. Yi, J. Hu and Z. Huang, *Energy Fuels*, 2024, **38**, 806–853.
- J. Low, J. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, *Adv. Mater.*, 2017, **29**, 1601694.
- J. Yang, D. Wang, H. Han and C. Li, *Acc. Chem. Res.*, 2013, **46**, 1900–1909.
- H. Zhao, L. Jian, M. Gong, M. Jing, H. Li, Q. Mao, T. Lu, Y. Guo, R. Ji and W. Chi, *Small Struct.*, 2022, **3**, 2100229.
- F. Zhang, Y. Zhu, Q. Lin, L. Zhang, X. Zhang and H. Wang, *Energy Environ. Sci.*, 2021, **14**, 2954–3009.
- G. Zhao and X. Xu, *Nanoscale*, 2021, **13**, 10649–10667.
- Z. Lin, L. Li, L. Yu, W. Li and G. Yang, *J. Mater. Chem. A*, 2017, **5**, 5235–5259.
- A. Kudo and Y. Miseki, *Chem. Soc. Rev.*, 2009, **38**, 253–278.
- C. Bie, L. Wang and J. Yu, *Chem*, 2022, **8**, 1567–1574.
- S. A. Ali, I. Sadiq and T. Ahmad, *J. Mol. Chem.*, 2025, **5**, 1204.
- A. K. Ghosh, U. Saha, S. Biswas, Z. A. Allothman, M. A. Islam and M. Dolai, *Ind. Eng. Chem. Res.*, 2022, **61**, 175–186.
- J. Xu, Q. Ji, X. Yan, C. Wang and L. Wang, *Appl. Catal., B*, 2020, **268**, 118739.
- B. Park, W.-W. Park, J. Y. Choi, W. Choi, Y. M. Sung, S. Sul, O.-H. Kwon and H. Song, *Chem. Sci.*, 2023, **14**, 7553–7558.
- R. Li, F. Zhang, D. Wang, J. Yang, M. Li, J. Zhu, X. Zhou, H. Han and C. Li, *Nat. Commun.*, 2013, **4**, 1432.
- T. Takata, J. Jiang, Y. Sakata, M. Nakabayashi, N. Shibata, V. Nandal, K. Seki, T. Hisatomi and K. Domen, *Nature*, 2020, **581**, 411–414.
- Y. Kageshima, H. Inuzuka, H. Kumagai, B. Ohtani, K. Teshima and H. Nishikiori, *J. Phys. Chem. C*, 2023, **127**, 18327–18339.
- L. Yuan, Z. Geng, J. Xu, F. Guo and C. Han, *Adv. Funct. Mater.*, 2021, **31**, 2101103.
- H. Liu, W. Xiong, C. Zhou, C. Lai, L. Li, G. Wang, X. Huo, G. Zeng and M. Cheng, *Coord. Chem. Rev.*, 2025, **529**, 216468.
- S. M. Wu and P. Schmuki, *Adv. Mater.*, 2025, **37**, 2414889.
- Y. Fu, K. Lu, A. Hu, J. Huang, L. Guo, J. Zhou, J. Zhao, O. V. Prezhdo and M. Liu, *J. Am. Chem. Soc.*, 2023, **145**, 28166–28175.
- H. Peng, T. Yang, H. Lin, Y. Xu, Z. Wang, Q. Zhang, S. Liu, H. Geng, L. Gu, C. Wang, X. Fan, W. Chen and X. Huang, *Adv. Energy Mater.*, 2022, **12**, 2201688.



- 45 Y. Akinaga, T. Kawawaki, H. Kameko, Y. Yamazaki, K. Yamazaki, Y. Nakayasu, K. Kato, Y. Tanaka, A. T. Hanindriyo, M. Takagi, T. Shimazaki, M. Tachikawa, A. Yamakata and Y. Negishi, *Adv. Funct. Mater.*, 2023, **33**, 2303321.
- 46 X. Ren, C. Li, J. Liu, H. Li, L. Bing, S. Bai, G. Xue, Y. Shen and Q. Yang, *ACS Appl. Mater. Interfaces*, 2022, **14**, 6885–6893.
- 47 Z.-Z. Liang, X.-A. Li, Q.-Z. Chen, X.-L. Wang, P.-Y. Su, J.-F. Huang, Y. Zhou, L.-M. Xiao and J.-M. Liu, *ACS Catal.*, 2024, **14**, 10447–10461.
- 48 J. Zhang, L. Li, M. Du, Y. Cui, Y. Li, W. Yan, H. Huang, X. A. Li and X. Zhu, *Small*, 2023, **19**, 2300402.
- 49 Q. Zhao, W. Yao, C. Huang, Q. Wu and Q. Xu, *ACS Appl. Mater. Interfaces*, 2017, **9**, 42734–42741.
- 50 Z. Zeng, Y. Su, X. Quan, W. Choi, G. Zhang, N. Liu, B. Kim, S. Chen, H. Yu and S. Zhang, *Nano Energy*, 2020, **69**, 104409.
- 51 T. Lv, B. Xiao, F. Xia, M. Chen, J. Zhao, Y. Ma, J. Wu, J. Zhang, Y. Zhang and Q. Liu, *Chem. Eng. J.*, 2022, **450**, 137873.
- 52 T. Kawawaki, Y. Mori, K. Wakamatsu, S. Ozaki, M. Kawachi, S. Hossain and Y. Negishi, *J. Mater. Chem. A*, 2020, **8**, 16081–16113.
- 53 C. Gao, F. Lyu and Y. Yin, *Chem. Rev.*, 2021, **121**, 834–881.
- 54 L. Zhang, X. Lu, J. Sun, C. Wang and P. Dong, *J. Mater. Chem. A*, 2024, **12**, 5392–5405.
- 55 B. Liu, Y. Li, Y. Guo, Y. Tang, C. Wang, Y. Sun, X. Tan, Z. Hu and T. Yu, *ACS Nano*, 2024, **18**, 17939–17949.
- 56 X. Xu, F. Luo, W. Tang, J. Hu, H. Zeng and Y. Zhou, *Adv. Funct. Mater.*, 2018, **28**, 1804055.
- 57 T. Raja Mogan, J. Zhang, L. S. Ng, S. K. Boong, C. Chong, J.-K. Lee, H. Li and H. K. Lee, *Angew. Chem., Int. Ed.*, 2024, **63**, e202401277.
- 58 X. Zhang, A. Fu, X. Chen, L. Liu, L. Ren, L. Tong and J. Ye, *Catal. Today*, 2019, **335**, 166–172.
- 59 H. Ren, J.-L. Yang, W.-M. Yang, H.-L. Zhong, J.-S. Lin, P. M. Radjenovic, L. Sun, H. Zhang, J. Xu, Z.-Q. Tian and J.-F. Li, *ACS Mater. Lett.*, 2021, **3**, 69–76.
- 60 J.-L. Yang, Y.-L. He, H. Ren, H.-L. Zhong, J.-S. Lin, W.-M. Yang, M.-D. Li, Z.-L. Yang, H. Zhang, Z.-Q. Tian and J.-F. Li, *ACS Catal.*, 2021, **11**, 5047–5053.
- 61 B. T. Sneed, A. P. Young, D. Jalalpoor, M. C. Golden, S. Mao, Y. Jiang, Y. Wang and C. K. Tsung, *ACS Nano*, 2014, **8**, 7239–7250.
- 62 S. Cao and J. Yu, *J. Phys. Chem. Lett.*, 2014, **5**, 2101–2107.
- 63 L. Zhang, F. Zhang, H. Xue, J. Gao, Y. Peng, W. Song and L. Ge, *Chin. J. Catal.*, 2021, **42**, 1677–1688.
- 64 Y. Xin, L. Wu, L. Ge, C. Han, Y. Li and S. Fang, *J. Mater. Chem. A*, 2015, **3**, 8659–8666.
- 65 K. U. Sahar, K. Rafiq, M. Z. Abid, U. U. Rehman, R. H. Althomali, A. Rauf and E. Hussain, *Energy Fuels*, 2024, **38**, 17995–18009.
- 66 K. Bhunia, M. Chandra, S. Khilari and D. Pradhan, *ACS Appl. Mater. Interfaces*, 2019, **11**, 478–488.
- 67 C. Wang, D. Drago, C. Colbeau-Justin, P. Haghi-Ashtiani, M. N. Ghazzal and H. Remita, *ACS Appl. Mater. Interfaces*, 2023, **15**, 42637–42647.
- 68 C. M. Pelicano, M. Saruyama, R. Takahata, R. Sato, Y. Kitahama, H. Matsuzaki, T. Yamada, T. Hisatomi, K. Domen and T. Teranishi, *Adv. Funct. Mater.*, 2022, **32**, 2202987.
- 69 E. Hussain, A. Ishaq, M. Z. Abid, M. Z. Waleed, A. Rauf, R. Jin and K. Rafiq, *ACS Appl. Energy Mater.*, 2024, **7**, 1914–1926.
- 70 V. Rout, B. Maji, H. V. Annadata, R. R. Maharana, D. K. Panda, J. Samantaray, U. K. Goutam, K. Samanta, M. Mishra and P. Dash, *ACS Appl. Mater. Interfaces*, 2024, **16**, 69333–69358.
- 71 Q. Zeng, Y. Bao, S. Ning, Q. Yu, Y. Wei and D. Zeng, *J. Mater. Chem. A*, 2024, **12**, 17286–17294.
- 72 H. Zhao, J. Wang, Y. Dong and P. Jiang, *ACS Sustainable Chem. Eng.*, 2017, **5**, 8053–8060.
- 73 Z. Liang, C. Yang, J. Lu and X. Dong, *Appl. Surf. Sci.*, 2021, **566**, 150732.
- 74 Q. Zhu, B. Qiu, H. Duan, Y. Gong, Z. Qin, B. Shen, M. Xing and J. Zhang, *Appl. Catal., B*, 2019, **259**, 118078.
- 75 S. Li, L. Wang, S. Liu, B. Xu, N. Xiao, Y. Gao, W. Song, L. Ge and J. Liu, *ACS Sustainable Chem. Eng.*, 2018, **6**, 9940–9950.
- 76 Q. Zhang, X. Wang, J. Zhang, L. Li, H. Gu and W.-L. Dai, *J. Colloid Interface Sci.*, 2021, **590**, 632–640.
- 77 S. Meng, P. An, L. Chen, S. Sun, Z. Xie, M. Chen and D. Jiang, *J. Colloid Interface Sci.*, 2021, **585**, 108–117.
- 78 C. Li, H. Wu, Y. Du, S. Xi, H. Dong, S. Wang and Y. Wang, *ACS Sustainable Chem. Eng.*, 2020, **8**, 12934–12943.
- 79 K. Li, X. Chen, J. Zhao, H. She, J. Huang, L. Wang and Q. Wang, *ACS Appl. Energy Mater.*, 2022, **5**, 10207–10215.
- 80 C. Li, H. Wu, S. Hong, Y. Wang, N. Song, Z. Han and H. Dong, *Int. J. Hydrogen Energy*, 2020, **45**, 22556–22566.
- 81 Q. Yue, Y. Wan, Z. Sun, X. Wu, Y. Yuan and P. Du, *J. Mater. Chem. A*, 2015, **3**, 16941–16947.
- 82 Z. Sun, Q. Yue, J. Li, J. Xu, H. Zheng and P. Du, *J. Mater. Chem. A*, 2015, **3**, 10243–10247.
- 83 H. Zhou, R. Chen, C. Han, P. Wang, Z. Tong, B. Tan, Y. Huang and Z. Liu, *J. Colloid Interface Sci.*, 2022, **610**, 126–135.
- 84 J. Zhang, W. Yao, C. Huang, P. Shi and Q. Xu, *J. Mater. Chem. A*, 2017, **5**, 12513–12519.
- 85 Z. Chen, Y. Bu, L. Wang, X. Wang and J.-P. Ao, *Appl. Catal., B*, 2020, **274**, 119117.
- 86 K. He, E. Campbell, Z. Huang, R. Shen, Q. Li, S. Zhang, Y. L. Zhong, P. Zhang and X. Li, *Small Struct.*, 2022, **3**, 2200104.
- 87 N. S. Powar, C. B. Hiragond, D. Bae and S.-I. In, *J. CO<sub>2</sub> Utiliz.*, 2022, **55**, 101814.
- 88 K. He, J. Xie, Z.-Q. Liu, N. Li, X. Chen, J. Hu and X. Li, *J. Mater. Chem. A*, 2018, **6**, 13110–13122.
- 89 Z. Zhao, J. Wu, Y.-Z. Zheng, N. Li, X. Li and X. Tao, *ACS Catal.*, 2019, **9**, 8144–8152.
- 90 W. Liu, D. Zhang, R. Wang, Z. Zhang and S. Qiu, *ACS Appl. Mater. Interfaces*, 2022, **14**, 31782–31791.
- 91 F. Yang, D. Liu, Y. Li, L. Cheng and J. Ye, *Chem. Eng. J.*, 2020, **399**, 125794.
- 92 J. Liu, P. Wang, J. Fan, H. Yu and J. Yu, *ACS Sustainable Chem. Eng.*, 2021, **9**, 3828–3837.
- 93 R. Tong, Z. Sun, X. Wang, S. Wang and H. Pan, *J. Phys. Chem. C*, 2019, **123**, 26136–26144.
- 94 Y. Dang, L. Feng, W. Hu, W. Wang, Q. Zhang and B. Ma, *Int. J. Hydrogen Energy*, 2021, **46**, 39251–39261.
- 95 L. Tian, S. Min and F. Wang, *Appl. Catal., B*, 2019, **259**, 118029.
- 96 Y. Peng, G. Han, D. Wang, K. Wang, Z. Guo, J. Yang and W. Yuan, *Int. J. Hydrogen Energy*, 2017, **42**, 14409–14417.
- 97 Q. Zhu, B. Qiu, M. Du, M. Xing and J. Zhang, *Ind. Eng. Chem. Res.*, 2018, **57**, 8125–8130.
- 98 S. Gupta, C. Prapaitrakool, B. R. Bhagat, C.-L. Yeh, A. Dashora, A. Sirisuk, N. Patel, N. Daneu, A. Kocjan, M. Spreitzer, J. C. S. Wu and M. M. Kržmanc, *Int. J. Hydrogen Energy*, 2024, **60**, 1288–1298.
- 99 A. Yu Kurenkova, A. D. Radina, V. S. Baidyshev, P. V. Povalyaev, E. E. Aidakov, E. Yu. Gerasimov, D. D. Mishchenko, A. V. Zhurenok, A. Y. Pak, E. A. Kozlova and A. G. Kvashnin, *Appl. Surf. Sci.*, 2024, **661**, 160095.
- 100 Z. Lv, Y. Wang, Y. Liu, J. Wang, G. Qin, Z. Guo and C. Zhang, *J. Phys. Chem. C*, 2022, **126**, 9041–9050.
- 101 L. Song, S. Zhang, D. Liu, S. Sun and J. Wei, *Int. J. Hydrogen Energy*, 2020, **45**, 8234–8242.
- 102 H. Long, P. Wang, X. Wang, F. Chen and H. Yu, *Appl. Surf. Sci.*, 2022, **604**, 154457.
- 103 L. Xiao, X. Li, J. Zhang and Z. He, *ACS Appl. Nano Mater.*, 2021, **4**, 12779–12787.
- 104 X. Li, S. Lu, J. Yi, L. Shen, Z. Chen, H. Xue, Q. Qian and M.-Q. Yang, *ACS Appl. Mater. Interfaces*, 2022, **14**, 25297–25307.
- 105 R. Shen, J. Xie, Y. Ding, S.-Y. Liu, A. Adamski, X. Chen and X. Li, *ACS Sustainable Chem. Eng.*, 2019, **7**, 3243–3250.
- 106 Z. Xia, C. Chen, X. Qi, Q. Xu, H. Tang and G. Liu, *Adv. Sustainable Syst.*, 2023, **7**, 2200134.
- 107 J. Guan, J. Wu, D. Jiang, X. Zhu, R. Guan, X. Lei, P. Du, H. Zeng and S. Yang, *Int. J. Hydrogen Energy*, 2018, **43**, 8698–8706.
- 108 Z. Lian, P. Xu, W. Wang, D. Zhang, S. Xiao, X. Li and G. Li, *ACS Appl. Mater. Interfaces*, 2015, **7**, 4533–4540.
- 109 S. Xu, M. Li, Y. Wang and Z. Jin, *Int. J. Hydrogen Energy*, 2024, **51**, 16–30.
- 110 P. Wang, P. Deng and Y. Cao, *Int. J. Hydrogen Energy*, 2022, **47**, 1006–1015.
- 111 H. Nishiyama, T. Yamada, M. Nakabayashi, Y. Maehara, M. Yamaguchi, Y. Kuromiya, Y. Nagatsuma, H. Tokudome, S. Akiyama, T. Watanabe, R. Narushima, S. Okunaka, N. Shibata, T. Takata, T. Hisatomi and K. Domen, *Nature*, 2021, **598**, 304–307.

