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Unveiling selectivity trends for CO_2 reduction reaction over $\text{Ti}_3\text{C}_2\text{T}_x$ MXene: the key role of less-stable intermediate states and coadsorbates

A new approach to derive selectivity trends based on theoretical calculations is developed, providing a new framework for the design of improved materials for energy conversion and storage processes using computational methods.

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The electrochemical conversion of carbon dioxide via the CO₂ reduction reaction (CO₂RR) is an attractive strategy for the production of value-added chemicals. However, the CO₂RR suffers from a selectivity problem due to the large number of carbon-based products that can be obtained and the competing hydrogen evolution reaction (HER). It has been experimentally shown that the ratio and chemical nature of terminal groups, T_x, present on the Ti₃C₂T_x (T_x = O, OH or F) surface under electrochemical conditions affect selectivity and activity trends of the MXene electrocatalyst. In the present manuscript, we use electronic structure theory calculations to comprehend the selectivity trends in the CO₂RR over Ti₃C₂T_x with different terminal groups, including *OH and *F adsorbates. We show that the traditional modeling approach used in calculations to derive activity and selectivity trends, which only includes the most stable intermediate state in the analysis, is not consistent with experimental observations. Rather, it is necessary to include energetically less favorable intermediate states and coadsorbates in the analysis of mechanistic pathways. Remarkably, the inclusion of less-stable intermediates, although stable on the electrode surface, and coadsorbates opens up new reaction channels that are energetically more favorable, and only by considering these extensions are we able to map our results to the experimental data. We believe that the reported finding is not only limited to the CO₂RR or MXene systems, but likely also plays an important role in other catalytic transformations under applied bias.

1 Introduction

The sustained combustion of fossil fuels required to fulfill the energy requests of our societies has led to a drastic increase in

Unveiling selectivity trends for CO₂ reduction reaction over Ti₃C₂T_x MXene: the key role of less-stable intermediate states and coadsorbates[†]

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New concepts

The present manuscript focuses on resolving selectivity trends for the CO₂RR over Ti₃C₂T_x MXene with different terminal groups, T_x, including *OH and *F adsorbates, using electronic structure theory calculations. The traditional approach in the community to understand selectivity in the CO₂RR involves constructing thermodynamic free-energy diagrams for the different reaction channels, where only the most stable intermediate states are considered in the mechanistic analysis. In the case of the CO₂RR over Ti₃C₂T_x MXene, this framework does not reproduce the experimentally observed selectivity trends. We demonstrate that less-stable intermediates and coadsorbates, which also lead to less-stable configurations, must be included in the analysis to determine selectivity trends that are consistent with experimental data. The inclusion of these new configurations, even if they are energetically less favorable, opens new reaction channels that are energetically more favorable and thus can change the activity and selectivity trends compared to the traditional approach. The presented concept of factoring less-stable intermediate states and coadsorbates into the mechanistic analysis based on adsorption free energies is not limited to the CO₂RR or MXene systems, and we believe that the proposed framework will guide the design of improved materials for energy conversion and storage processes using computational methods.

the anthropogenic CO₂ levels, causing devastating effects on the environment and life of many ecosystems. Although a transition to green energies would be desirable, it is still not feasible to cover all energy demands exclusively with green energies.¹ CO₂ valorization emerges as an appealing strategy to counteract the current situation and represents a carbon neutral route for the generation of value-added energy carriers, while simultaneously reducing the anthropogenic CO₂ levels.²

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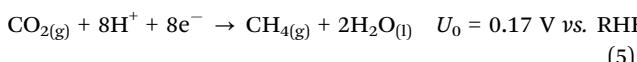
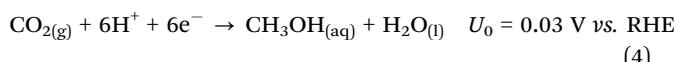
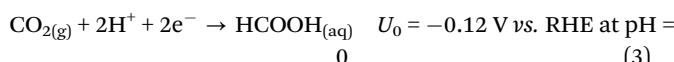
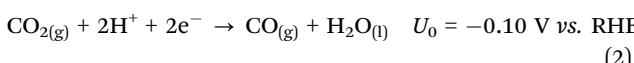
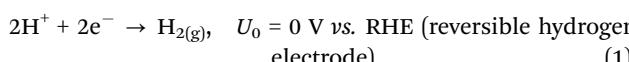
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[†] Electronic supplementary information (ESI) available: Computational details, elementary steps for CO₂RR and HER, calculation of formation Gibbs free energies, gas-phase error corrections, CO₂RR to HCOOH under different pH conditions, activity descriptor $G_{\max}(U)$, free energy diagrams and potential dependency for CO₂RR and HER for the different Ti₃C₂T_x surface models for different assumptions, negative effect of including less-stable intermediate states, rotational energy barriers, and schematic representation of CO₂RR to CO on the neighboring site of a preadsorbed CO. Relevant inputs and outputs for the different calculations have been also made available a public GitHub repository: https://github.com/plozanore/CO2RR_selectivity_key_role_of_less-stable_intermediate_states. See DOI: <https://doi.org/10.1039/d5mh00168d>



Currently, different strategies are used for CO₂ conversion, which include thermo,^{3–10} photo,^{11–14} and electrocatalysis,^{15–23} among others. Interestingly, CO₂ conversion *via* the electrocatalytic CO₂ reduction reaction (CO₂RR) is a very interesting process as it is carried out under mild operating conditions, usually at room temperature and moderate pressures. In addition, renewable energy resources can provide the required electricity for electrolysis, making the electrochemical pathway also environmentally attractive.

The electrochemical CO₂RR is mainly limited by its selectivity challenge. On the one hand, the formation of H_{2(g)} from the hydrogen evolution reaction (HER) (*i.e.*, eqn (1)) is observed for cathodic polarization, which is a competing side reaction under CO₂RR conditions.²⁴ On the other hand, there is the possibility of forming multiple C₁ products (*i.e.*, eqn (2)–(5)) or multi-carbon products (*i.e.*, C₂₊).²⁵



While the formation of C₂₊ products in the CO₂RR is of greater importance than the formation of C₁ products, the former products are normally obtained only from Cu-based catalysts, while CO, HCOOH and H₂ are typically observed for other transition metals.²⁵ Interestingly, the emergence of a new class of two-dimensional transition-metal carbides and nitrides, so-called MXenes, with excellent properties for energy conversion and storage, opens new avenues for the development of next-generation catalysts with improved catalytic activity and/or selectivity.²⁶ In this regard, it has been recently shown that the Cu-doped Ti₃C₂T_x (T_x = O, OH or F) MXene or even its basal planes can produce different C₁ products and even C₂₊ species for the CO reduction reaction (CORR) and CO₂RR.^{27,28} These findings suggest that terminal groups can modulate the CO₂RR selectivity and pave the way for the development of MXenes-based catalysts for selective CO₂RR.

An atomistic understanding of the mechanism that governs the reaction is essential for the rational design of active and selective catalysts. Interestingly, such understanding can be achieved from electronic structure calculations, usually in the density functional theory (DFT) framework, by materials screening. Such approach offers the opportunity to investigate and identify potential candidate materials that can later be experimentally validated,^{29,30} which may help overcoming the CO₂RR selectivity problem. The traditional approach used in DFT to study electrocatalytic transformations focus on the

thermodynamic picture. This is justified by the consideration of Brønsted–Evans–Polanyi relation,^{31–33} which connects the kinetics and thermodynamics. More precisely, the traditional approach comprises to calculate different configurations of the reaction intermediates in the CO₂RR or other electrocatalytic processes to identify the thermodynamically most stable structures.^{34–38} Thereafter, free-energy diagrams are constructed for these stable reaction intermediates, which allow predicting activity and selectivity trends of materials using descriptor-based analyses.^{39,40}

In this contribution, we aim to introduce a new framework for constructing and analyzing free-energy diagrams to enable a thorough discussion of activity and selectivity trends. In our approach, we factor not only the thermodynamically most stable reaction intermediates but also the less-stable reaction intermediates into the analysis of the elementary reaction steps. We illustrate our methodology using the CO₂RR over the Ti₃C₂T_x MXene with different *OH/*F adsorbates as terminal groups and discuss the role of less-stable intermediate states and coadsorbates, which also give rise to less-stable configurations, on the activity and selectivity. We demonstrate that less-stable intermediate states give rise to energetically favored reaction channels compared to the common assumption of considering the thermodynamically most stable structures only. Interestingly, our predicted activity and selectivity trends are only in good agreement with experimental data if less-stable intermediate states and coadsorbates are considered in the analysis, whereas the traditional approach, focusing on the thermodynamically most stable reaction intermediates only, fails to reproduce the experimentally reported trends.

2 Computational methods

In this study, we apply periodic DFT calculations as implemented in the Vienna Ab initio Simulation Package (VASP) code^{41–43} to model the CO₂RR and HER on different (3 × 3) Ti₃C₂T_x (T_x = *OH and/or *F) surface models. All calculations have been done using the Perdew–Burke–Ernzerhof (PBE) exchange correlation functional⁴⁴ alongside Grimme’s D3 term to account for dispersion effects.⁴⁵ We use the computational hydrogen electrode (CHE) approach⁴⁶ to describe the energetics of proton–electron transfer steps occurring during CO₂RR and HER. Transition states (TS) have been located using the Catlearn Bayesian transition state search module (ML-NEB).⁴⁷ All computational details are provided in Section S1 of the ESI.†

3 Surface models and reaction network

Recently Krishnan *et al.*,²⁸ studied the CO₂RR using linear sweep voltammetry (LSV) and chronoamperometry (CA) techniques on the basal plane of the Ti₃C₂T_x (T_x = *O, *OH, or *F) MXene, which contain a considerable amount of *F adsorbates. Interestingly, they observed suppression of the HER and high selectivity towards CO and CH₃OH, while also finding



considerable amounts of C_{2+} species (*i.e.*, ethanol and acetone). Bao *et al.*²⁷ studied the CORR on Cu-based $Ti_3C_2T_x$ systems by combining experimental and theoretical approaches. In their LSV experiments, they observed the formation of considerable amounts of ethanol and ethylene for the Cu-based $Ti_3C_2T_x$ system. However, they found high HER selectivity when the CORR was studied on the basal plane of $Ti_3C_2T_x$, with a large fraction of *O and *OH terminal groups. Note that Pourbaix diagrams for the Ti_3C_2 MXene under cathodic conditions predict that the surface is largely covered by $^*OH/^*O$ adsorbates, and *F -covered surfaces are found to be stable at low pH.^{48,49}

Nevertheless, a larger *F coverage can also be found at different working conditions due to the synthetic conditions used during the MXene preparation.²⁸ The above findings suggest that the *F content can play a significant role in CO_2RR selectivity. Notably, in line with the above results, the important role of surface terminations for Ti_3C_2 MXene on the catalytic HER activity was reported.⁴⁹ To analyze this effect, we study the CO_2RR on different $Ti_3C_2T_x$ MXene surface models with different ratios of $^*OH/^*F$ adsorbates (see Fig. 1).

To study the CO_2RR and HER on the different surface models, we have considered two different reaction networks. The first reaction network considers the reaction intermediates in their thermodynamically most stable configuration (Fig. 2a) and comprises a total of 18 adsorbates, 7 gas-phase molecules, and 30 elementary steps. For the second reaction network, we have also included other less-stable intermediate states that better connect two intermediate states and provide a more realistic mechanistic description (Fig. 2b). This gives rise to a total of 23 intermediate states, 7 gas-phase molecules, and 41 elementary steps considered in the analysis. Let us emphasize that a proper representation of the 'natural pathway' of product formation (without assuming any free internal rotation) can only be achieved if less-stable intermediate states are factored into the mechanistic description. All elementary steps of the CO_2RR and HER can be found in Section S2 of the ESI[†] (*i.e.*, eqn (S5)–(S34)).

To derive activity and selectivity trends for the CO_2RR and HER over $Ti_3C_2T_x$, we have used the following procedure. First, we calculated the adsorption energy of the different intermediate species for the different surface sites shown in the inset of Fig. 1 and determined the energetically preferred surface site corresponding to the fcc_{in} site. Note that for adsorbates that can adopt multiple configurations, we performed different geometrical optimizations and used the geometry of the most stable state (Fig. 2a), but also some geometries of the less-stable states (Fig. 2b). Second, we have also located the transition state (TS) for rotations between the less and most stable states and calculated the corresponding rotational barriers. Third, frequency calculations were performed to confirm that all configurations correspond to real minima or TS. In addition, frequency calculations are required to account for the zero-point energy and entropic contributions, which are needed to derive the Gibbs free energy of the different adsorbates. We refer the reader to Section S1.1 (ESI[†]) for a detailed description of how Gibbs free energies are derived. Fourth, we use the CHE approach⁴⁶ to describe the energetics of proton–electron transfer steps occurring during the CO_2RR and HER and construct free-energy diagrams for the different intermediate states. Fifth, the free-energy diagrams are analyzed using a descriptor-based approach^{50–52} to derive activity and selectivity trends. Finally, the predicted trends are compared to the experimental data of Krishnan *et al.*²⁸

4 Results

Let us start by analyzing how the different adsorbates interact with the different $Ti_3C_2T_x$ surfaces (*cf.* Fig. 1), as shown in Fig. 3. The interaction is calculated from the formation Gibbs free energy, which is derived using $H_{2(g)}$, $CO_{2(g)}$, $H_2O_{(g)}$, and the bare $Ti_3C_2T_x$ surfaces as references. For a detailed description of how formation Gibbs free energies are calculated, the reader is referred to Section S3 in the ESI.[†] Note that the more/less negative the formation Gibbs free energy, the stronger/weaker the interaction between the adsorbate and the surface.

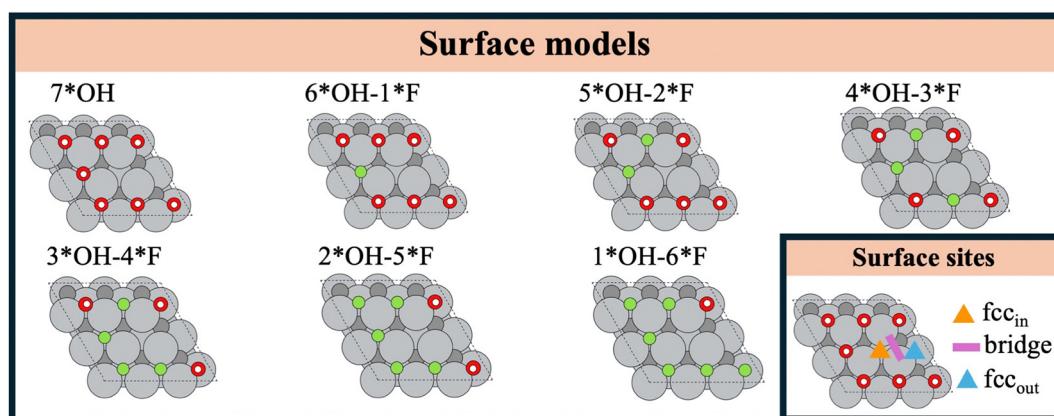


Fig. 1 Representation of the seven $Ti_3C_2T_x$ ($T_x = OH$ or F) surface models considered for the study of the CO_2RR and HER as a function of the *OH and *F terminal groups. Three different surface sites (*i.e.*, fcc_{in} , bridge and fcc_{out}) are considered as possible initial configurations for the adsorbates. Light grey, dark grey, red, white, and green balls denote Ti, C, O, H, and F atoms, respectively.



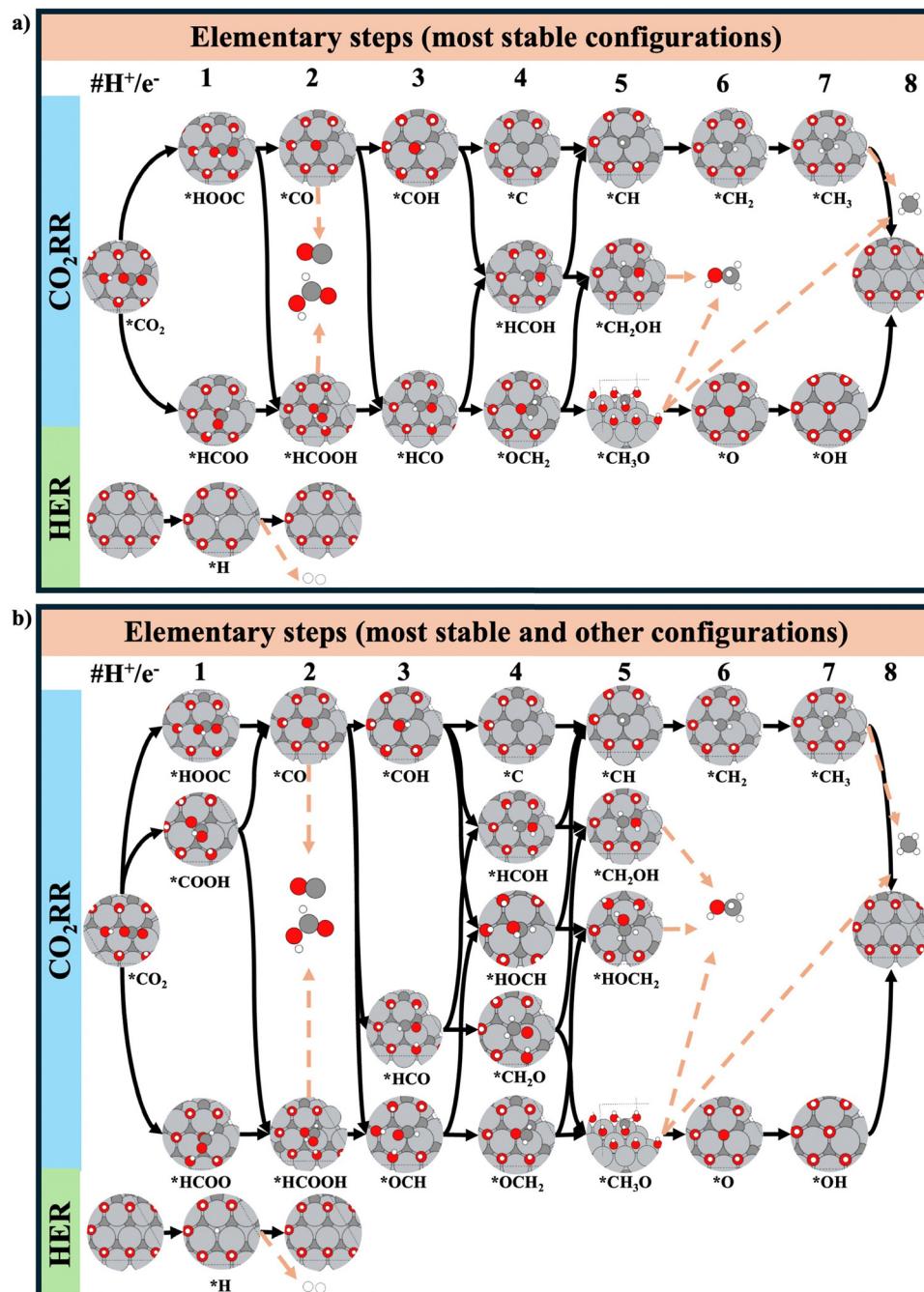


Fig. 2 Schematic representation of the elementary steps considered for the CO₂RR and HER on the example of the Ti₃C₂–*7OH surface: (a) only the most stable configuration of a reaction intermediate is considered, (b) besides the most stable configuration of an adsorbate, also less-stable intermediate states are taken into account. Note that we have considered similar adsorbate configurations for the other surface models studied. For the CO₂RR, four different C₁ products have been factored into the analysis, namely: CO, HCOOH, CH₃OH, and CH₄. Black arrows represent connections between two adsorbed intermediates, while brown arrows represent product formation. Light grey, dark grey, red and white spheres denote Ti, C, O and H atoms, respectively. All elementary steps are listed in eqn (S5)–(S34) of the ESI† (Section 2).

From Fig. 3, it appears that *H interacts similarly with all the surfaces, which implies that the HER is not largely dependent on the coverage of *F adsorbates. On the other hand, the larger the *F content, the weaker the *CO₂ interaction with the surface. As *CO₂ can limit the reaction at low cathodic potentials (*vide infra*), a more exergonic CO₂ adsorption is likely to limit

the CO₂RR activity. Therefore, surfaces with large *F content are likely to increase the CO₂RR activity. Regarding CO and HCOOH (*i.e.*, two possible products), the higher the *F coverage, the weaker the CO interaction with the surface, while the interaction of HCOOH remains almost the same. This suggests a positive effect of *F composition on the CO selectivity

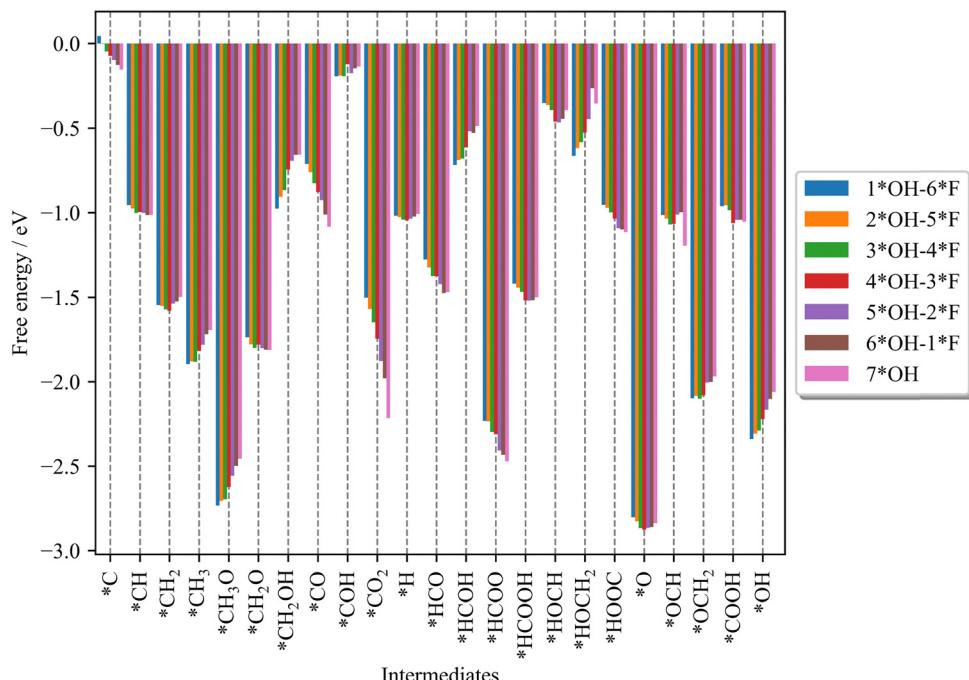


Fig. 3 Formation Gibbs free energy for the different reaction intermediates in the CO₂RR and HER on the different surfaces (cf. Fig. 1). A description of the calculation of the formation Gibbs free energies can be found in Section S3 of the ESI.†

compared to HCOOH. Until now, we have shown that some trends can be derived for reactants (CO₂) and some products (CO and HCOOH) as a function of *F/*OH coverages. However, no clear trends are observed for the intermediate states (Fig. 2). Therefore, free-energy diagrams for the CO₂RR and HER for the different surface models are constructed to gain insights into activity and selectivity.

5 Discussion

Let us start this section by defining some important aspects: (i) for the discussion of our results, we have selected the Ti₃C₂-4*OH-3*F surface model unless otherwise stated. The choice for this surface is because it has a similar *F/*OH composition. (ii) The potential used in our computational setup to construct the free-energy diagrams is $U = -0.4$ vs. RHE, which is chosen to mimic the potential of -1.1 V vs. Ag/AgCl at pH ≈ 7 –8 used in the experiments.²⁸ (iii) Gas-phase error corrections have been incorporated to correctly describe the overall thermodynamics on the free-energy diagrams.^{53,54} Further details on gas-phase error corrections can be found in Section S4 of the ESI.† (iv) For aqueous (*i.e.*, HCOOH and CH₃OH) and ionic species (*i.e.*, HCOO[−]), solvation and ion corrections have been applied, respectively.⁵⁵ Note that no changes are observed when using either HCOOH or HCOO[−] to assess the selectivity for formic acid/formate formation. Therefore, for simplicity, we have considered HCOOH instead of HCOO[−] in our analysis. For a detailed discussion, the reader is referred to Section S5 of the ESI.† Let us finally point out that, while pH effects can play an important role in modulating catalytic activity,^{56,57} in this study

we are interested in activity and selectivity trends as a function of terminal groups at the experimental working conditions.²⁸

To assess the activity and selectivity for the CO₂RR and HER, we make use of the descriptor $G_{\max}(U)$.^{50,51} This activity descriptor quantifies the largest free-energy span between reaction intermediates of a given mechanism at a given potential (U). The free-energy spans are calculated based on the potential-dependent Gibbs free energy of the different intermediates ($G_i(U)$), which are derived following eqn (S4) in Section S1.1 of the ESI.† For a detailed description of $G_{\max}(U)$ in the CO₂RR, the reader is referred to Section S6 of the ESI.† When evaluating multiple pathways, the mechanism with the lowest $G_{\max}(U)$ value is considered the most plausible mechanism. Interestingly, $G_{\max}(U)$ can incorporate several elementary steps in the analysis, making it a particularly useful descriptor for reactions with a high number of proton–electron transfer steps such as CO₂RR. Note that the lower the $G_{\max}(U)$, the higher the catalytic activity. In our analysis, we extract the lowest $G_{\max}(U)$ among the different possible mechanism that connect reactants to products (Fig. 2). This evaluation is performed for all products (*i.e.*, H₂, CO, HCOOH, CH₃OH and CH₄) across all different surface models (Fig. 1). Then, based on the $G_{\max}(U)$ values of the different reaction channels, we derive selectivity trends.

The free-energy diagram for CO₂RR, HER and the potential dependency of both reactions for the Ti₃C₂-4*OH-3*F surface model at $U = -0.4$ V vs. RHE is shown in Fig. 4. Similar diagrams for the other surfaces can be found in Section S7 of the ESI† (Fig. S5–S10).

Among the different C₁ products that can be produced during CO₂RR, CH₄ and CH₃OH are the most selectively produced products as evidenced by their lowest $G_{\max}(U)$ value

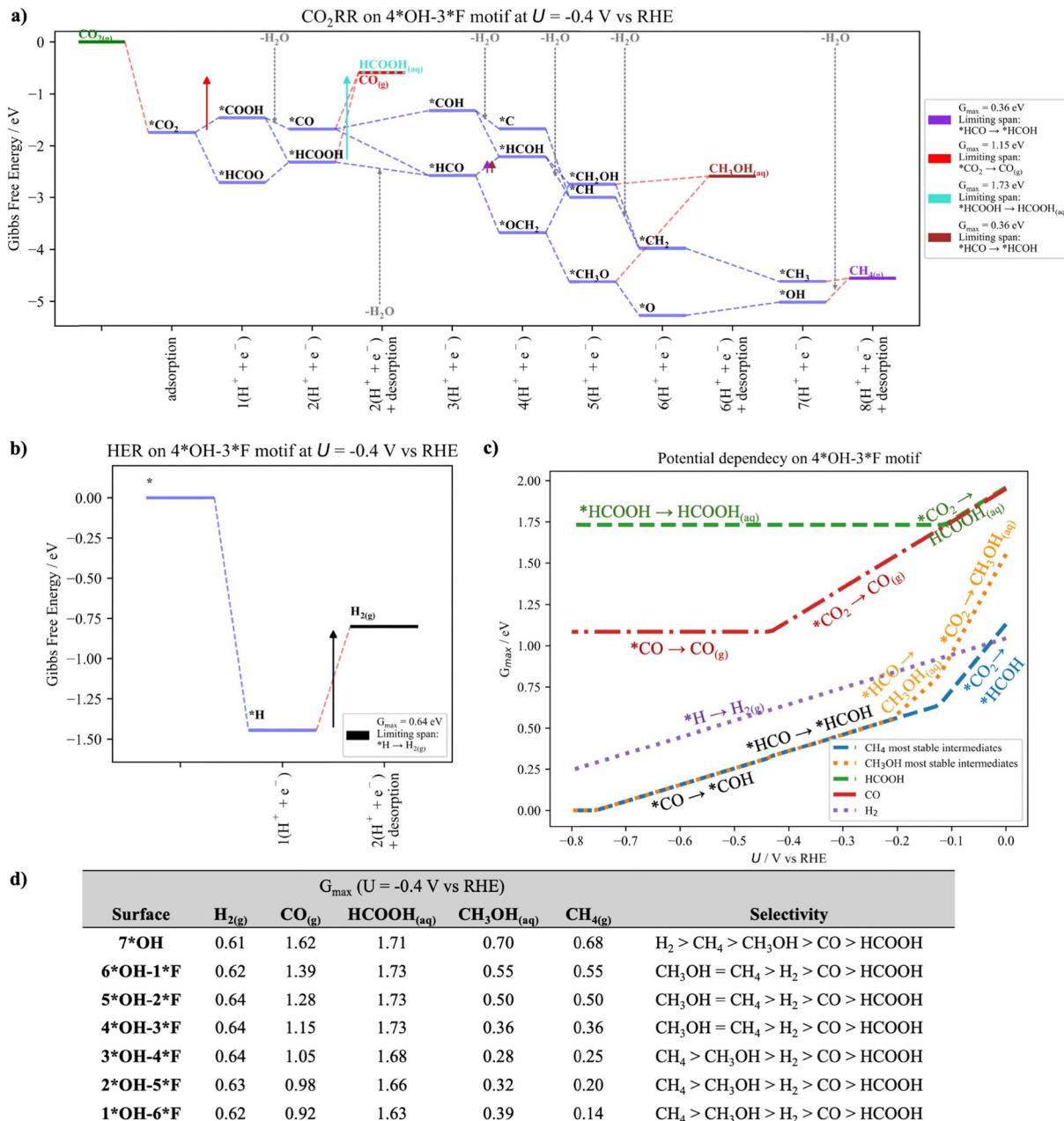


Fig. 4 Gibbs free energy diagram for the $\text{Ti}_3\text{C}_2\text{-}4\text{*OH-}3\text{*F}$ surface model at $U = -0.4$ V vs. RHE for (a) CO_2RR and (b) HER. Purple, red, cyan, brown, and black arrows are used to schematically represent the limiting span for $\text{CH}_4\text{(g)}$, $\text{CO}\text{(g)}$, $\text{HCOOH}_{\text{(aq)}}$, $\text{CH}_3\text{OH}_{\text{(aq)}}$ and $\text{H}_2\text{(g)}$, respectively. Details on the descriptor G_{\max} and intermediates involved in the limiting span for each reaction are shown in the legend. (c) CO_2RR and HER potential dependency of the $\text{Ti}_3\text{C}_2\text{-}4\text{*OH-}3\text{*F}$ surface model. Green, red, orange, blue, and purple lines represent the G_{\max} value as a function of the applied cathodic potential (U) for the formation of $\text{HCOOH}_{\text{(aq)}}$, $\text{CO}\text{(g)}$, $\text{CH}_3\text{OH}_{\text{(aq)}}$, $\text{CH}_4\text{(g)}$ and $\text{H}_2\text{(g)}$, respectively. The different labels represent the species involved in the limiting span for each reaction at each condition. Note that for two reactions comprising the same limiting span, black labels are used. (d) Selectivity and $G_{\max}(U = -0.4$ V vs. RHE) for the different products on the different surface models.

(Fig. 4a). Regarding CO and HCOOH, the stronger interaction of $^*\text{HCOOH}$ with the surface compared to that of $^*\text{CO}$ and $^*\text{CO}_2$ hinders its formation, making HCOOH the least selectively produced product. Interestingly, at $U = -0.4$ V vs. RHE, the HER (Fig. 4b) is suppressed in favor of CO_2RR because its $G_{\max}(U)$ value is larger than that for the formation of CH_3OH and CH_4 . Importantly, the formation of CH_3OH and CH_4

comprise many proton-electron transfer steps, and their limiting span in the approximation of $G_{\max}(U)$ consists of several elementary reaction steps. On the other hand, the G_{\max} for HER consists of a single step (*i.e.*, $^*\text{H} \rightarrow \text{H}_2\text{(g)}$). This implies that $G_{\max}(U)$ for CO_2RR is reduced more efficiently than for HER when switching to more cathodic potentials (Fig. 4c). Nevertheless, HER is still favored at low cathodic potentials (Fig. 4c).

This feature for HER is maintained for the MXene surfaces with a *F composition equal to or less than $3{}^*F$ (see Fig. S5–S10, ESI \dagger). This is because the larger the *F coverage, the weaker the *CO_2 interaction with the surface, while the *H interaction is almost the same for all surfaces (Fig. 3). Since *CO_2 is encountered in the limiting span at low cathodic potentials, the lower the CO_2 adsorption, the lower the $G_{\max}(U)$ and the higher the selectivity toward the CO_2 RR.

Let us now analyze the selectivity toward CO_2 RR and HER at $U = -0.4$ vs. RHE for all surfaces. From Fig. 4d three interesting patterns can be observed: (i) the highest *OH content favors HER, while increasing the *F content favors CO_2 RR, in agreement with experiments.^{27,28} This is because the reaction intermediates of CO_2 RR interact differently depending on the surface composition, while the *H interaction is similar for all surfaces. (ii) The highest selectivity is observed for CH_3OH and CH_4 at low to medium *F coverage and switches to CH_4 at larger *F composition. The latter is in contrast to experiments in which CO and CH_3OH are produced, but no CH_4 .²⁸ (iii) The lowest selectivity is towards CO and $HCOOH$. While the latter product is not observed in experiments, the former is the most selectively produced product for MXene surfaces at large *F coverages.²⁸ From these results, it is clear that the traditional model based on the analysis of the thermodynamically most stable intermediate states can only partially predict the experimental observations. In the following sections, we discuss the relevance of including less-stable intermediates states and coadsorbates for a refined mechanistic description to derive selectivity trends that are consistent with the experimental results.

5.1 Effect of including less-stable intermediate states

In this section, we discuss the importance of considering less-stable intermediate states in the activity analysis. There are two main reasons why this is important: (i) the configuration of the two most stable states of two consecutive intermediates may not be well connected. For instance, in Fig. 2a, *HCOOH and *HCO reveal a different orientation on the MXene surface: to connect these intermediate states, not only a proton–electron transfer but also an internal rotation is needed. The tacit assumption that adsorbed species can rotate freely with a small energy penalty may not hold true, especially for systems with high coverage. This situation is particularly evident in MXenes, where multiple terminal groups cover the surface and thus rotations from one intermediate state to another can be difficult (*vide infra*). Therefore, for a better mechanistic description, it appears important to consider configurations that truly connect two intermediate states, even if these new configurations correspond to energetically less-stable intermediates. This allows a more accurate and better representation of the ‘natural pathway’ (without free internal rotation) of a given mechanism. Note that this can also change the predicted activity to lower activities. (ii) By including less-stable intermediate states in the analysis, the number of possible pathways from the reactants to products increases. This enhances the possibilities of finding new pathways with a lower energy and thus higher catalytic

activity. Note that we do not consider different sites, but rather the same site with different configurations for the adsorbate.

Fig. 5a depicts the free-energy diagram for CO_2 RR including also less-stable intermediate states (marked in light blue). Similar figures for the other surfaces can be found in Section S8 of the ESI \dagger (Fig. S11–S16). Interestingly, there is a decrease in the $G_{\max}(U)$ and a change of the limiting span for CH_3OH and CH_4 compared to the situation where only the most stable intermediate states are considered (Fig. 4a). Let us focus on the example of CH_3OH and discuss the differences. The limiting span when the most stable states are considered is ${}^*HCO \rightarrow {}^*HCOH$, which changes to ${}^*CO_2 \rightarrow {}^*COOH$ if also less-stable intermediate states are considered. The inclusion of less-stable intermediate states (*i.e.*, *OCH , *HOCH and *HOCH_2) opens up new, energetically favorable pathways through these intermediate states. Interestingly, the formation of *OCH from *CO is still exergonic, and the free energy required to form *HOCH from *OCH is lower than the free energy required for the transition of *HCO to *HCOH . Finally, both the formation of *HOCH_2 and $CH_3OH_{(aq)}$ are downhill in free energy. This situation changes the limiting span to ${}^*CO_2 \rightarrow {}^*COOH$ and reduces the $G_{\max}(U)$ while increasing the catalytic activity. Note that, while including less-stable intermediate states can lead to new pathways with improved catalytic activity, also situations with lower catalytic activity can be encountered. In particular, we found this situation for the formation of CH_3OH on the $Ti_3C_2-7{}^*OH$ surface model (see Fig. S11b, ESI \dagger) in the potential range of $(-0.55 < U < -0.45$ V vs. RHE), which is a result of the rigorous connection of two consecutive intermediates, namely *HCOOH and *OCH (*or* *HCO). A detailed description can be found in Section S9 of the ESI \dagger .

While we have shown above that including less-stable intermediate states can lead to new reaction channels with improved catalytic activity, it is still not clear whether the less-stable intermediate states are truly accessible during catalytic operation. To gain deeper insights into their stability and feasibility as new reaction channels, we have calculated the rotational free-energy barrier between the less and most stable intermediate states, as summarized in Table S3 of Section S10 of the ESI \dagger . The rotational energy barrier is then compared with the energy required for the competing protonation reactions, as shown in Fig. 5b for the *OCH and *HCO intermediates as a representative example. Before analyzing Fig. 5b, let us first discuss the accessibility of the *OCH and *HCO intermediates. On the hand, *OCH and *HCO can be formed after *CO protonation, both being exergonic processes (Fig. 5a). On the other hand, if no free internal rotations are assumed, only *OCH —the less-stable intermediate state—can be formed after *HCOOH protonation, with a small free-energy penalty of 0.14 eV. This suggests that the formation of *OCH is more probable and therefore more accessible than *HCO . Nevertheless, the stability of the less-stable intermediate state is still unclear and questions whether *OCH will transition to *HCO or protonate instead.

As shown in Fig. 5b, the rotation of *OCH to *HCO and the reverse process (green region) have a free-energy barrier of 0.51



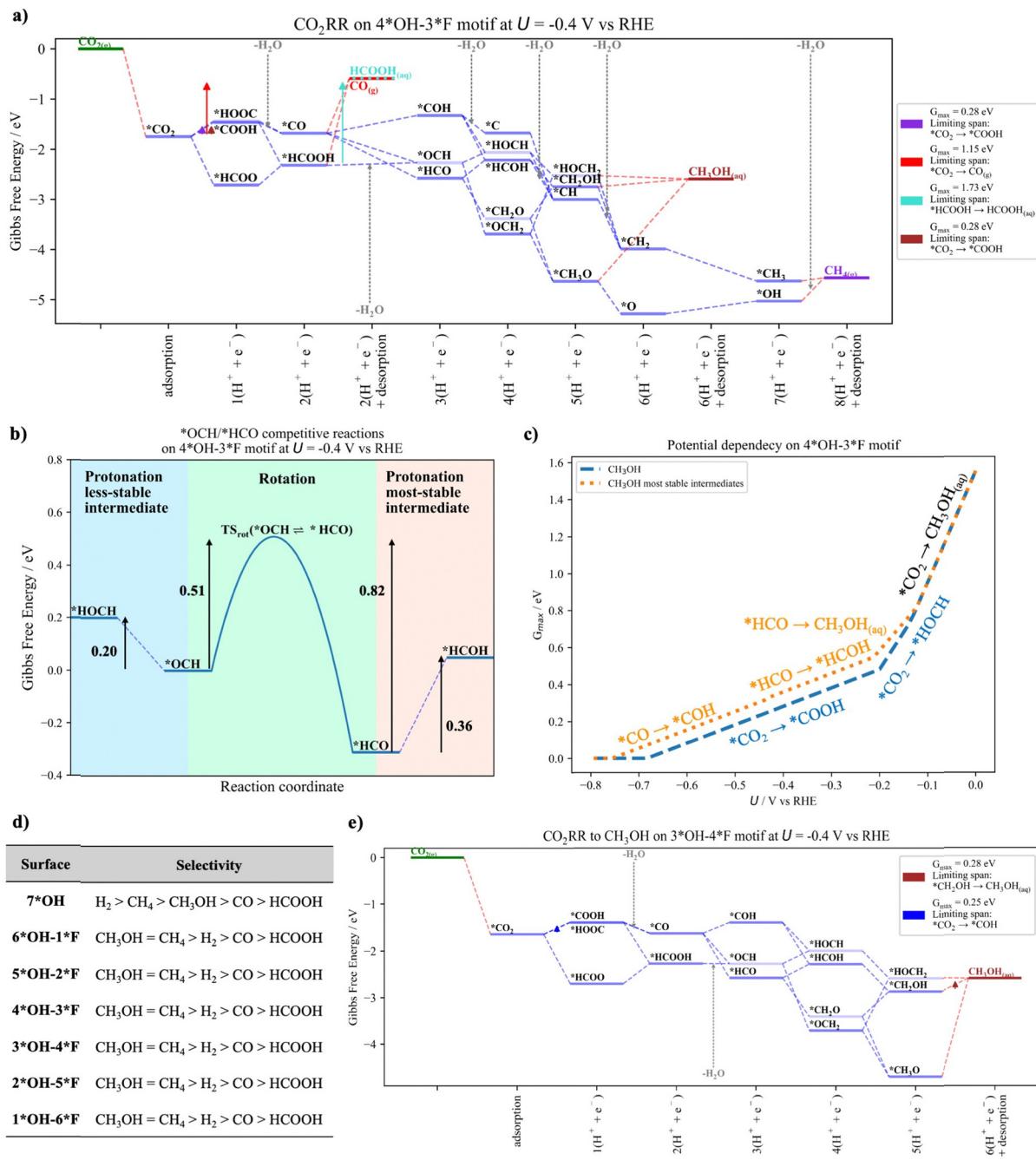


Fig. 5 (a) Gibbs free energy diagram for CO_2 RR of the $Ti_3C_2-4*OH-3*F$ surface at $U = -0.4$ V vs. RHE considering less-stable intermediates. Purple, red, cyan, and brown arrows are used to schematically represent the limiting span for the CO_2 RR to $CH_{4(g)}$, $CO_{(g)}$, $HCOOH_{(aq)}$, and $CH_3OH_{(aq)}$, respectively. For intermediates in which two different states are considered, lighter bars represent the less-stable intermediate while darker bars represent the most stable intermediate. (b) $*OCH$ and $*HCO$ competitive reactions of the $Ti_3C_2-4*OH-3*F$ surface at $U = -0.4$ V vs. RHE. (c) Potential dependency of CO_2 RR to CH_3OH of the $Ti_3C_2-4*OH-3*F$ with(out) considering less-stable intermediate states in orange and blue, respectively. The different labels represent the species involved in the limiting span for each reaction at each condition. Note that for two reactions comprising the same limiting span, black labels are used. (d) Selectivity trends for the different surface models at $U = -0.4$ V vs. RHE. (e) Gibbs free energy diagram for CO_2 RR to CH_3OH of the $Ti_3C_2-3*OH-4*F$ surface at $U = -0.4$ V vs. RHE. Blue and brown arrows represent the limiting span when less-stable intermediate states are considered or not, respectively.

and 0.82 eV, respectively, while their protonation to $*HOCH$ (blue region) and $*HCOH$ (salmon region) are endergonic processes with a reaction energy of 0.20 and 0.36 eV, respectively. From Fig. 5b it is evident that the rotation from one state

to another requires a higher energy barrier (kinetics) compared to the thermodynamic barrier for their protonation, which points to a high stability of the intermediate and suggests that the intermediate will protonate rather than rotate to the other

state. Note that for a better comparison, the knowledge of the free-energy barrier for the protonation step is required. While we have not explicitly calculated the protonation energy barriers, they are known to be small (0.15 eV to 0.25 eV).^{58,59} Particularly, the $^*\text{HCO}$ protonation to $^*\text{HCOH}$ is a process with a small free-energy barrier of 0.16 eV for the Cu(100) surface.⁶⁰ Therefore, if these values are considered (and added to the reaction energy if necessary), the protonation step is also kinetically more favorable (lower free-energy barrier) than the rotation, which suggests that the intermediate will prefer to protonate rather than rotate. Finally, it is important to consider that protonation steps are potential dependent while rotations are not. Therefore, at large cathodic overpotentials, the protonation step is clearly preferred over rotation to the more stable intermediate. Similarly, a high stability of the less-stable intermediate states is also observed for the other surfaces and less-stable intermediates states (*i.e.*, $^*\text{HOCH}$ and $^*\text{HOCH}_2$), with the only exception of $^*\text{OCH}$ and $^*\text{HOCH}_2$ on the 7^*OH and $6^*\text{OH}-1^*\text{F}$ surfaces, as summarized in Section S10 of the ESI.[†] Overall, these results point to a high stability of the less-stable intermediate states and their feasibility as new reaction channels.

The tacit assumption used in previous works^{34–38} that the most thermodynamically stable configurations of two consecutive reaction intermediates are well connected may not be true. For an improved mechanistic description, reaction intermediates must be well connected to each other and must not exhibit internal rotations, especially for those systems with high coverage. This implies that energetically less-stable intermediate states have to be considered in the analysis, which can lead to increased or reduced catalytic activity for a particular reaction channel.

Fig. 5c illustrates the potential dependency of CO_2RR to CH_3OH when the most stable (and also less-stable) intermediate states are considered in the analysis. The effect of incorporating less-stable intermediate states in the analysis is rapidly observed (*i.e.*, at $U < -0.15$ V vs. RHE) and is maintained under larger cathodic potentials. Initially, the limiting span (*i.e.*, $^*\text{CO}_2 \rightarrow \text{CH}_3\text{OH}_{(\text{aq})}$) is limited by the strong $^*\text{CO}_2$ adsorption and thus no effects are observed. At medium cathodic potentials (*i.e.*, $-0.45 < U < -0.15$ V vs. RHE), the limiting span for the most stable intermediates is either governed by the presence of the intermediate states $^*\text{HCO}$ or $^*\text{HCOH}$. In contrast, the consideration of the less-stable intermediate states switches the mechanism to an energetically favorable path, which contains neither $^*\text{HCO}$ nor $^*\text{HCOH}$. Finally, at large cathodic potentials (*i.e.*, $U < -0.45$ V vs. RHE), the limiting span for the most stable intermediate states is $^*\text{CO} \rightarrow ^*\text{COH}$. While in this case no other less-stable configurations are considered for $^*\text{CO}$ and $^*\text{COH}$, the $^*\text{CO}$ intermediate can form $^*\text{OCH}$, which then evolves to CH_3OH via a less energy-intensive pathway, and the limiting span switches to $^*\text{CO}_2 \rightarrow ^*\text{COOH}$.

Let us now evaluate the effect of including less-stable intermediate states on the CO_2RR selectivity for the different surface models at working conditions (*i.e.*, $U = -0.4$ V vs. RHE). As visible in Fig. 5d, H_2 is still the most selectively produced product for the $\text{Ti}_3\text{C}_2-7^*\text{OH}$ surface as observed in experiments

with large $^*\text{OH}$ coverage.²⁷ With increasing $^*\text{F}$ coverage, CH_3OH and CH_4 become the more selectively produced products, which is in better agreement with the experiments at high $^*\text{F}$ content,²⁸ where the most selectively produced products are CO and CH_3OH . Interestingly, only when including also less-stable intermediate states in our analysis, we can predict a high selectivity toward CH_3OH compared to the traditional approach (Fig. 4d). We suggest that this change of selectivity is a result of the different $^*\text{CH}_2\text{OH}$ and $^*\text{HOCH}_2$ interactions with the surfaces, as exemplified in Fig. 5e for the CO_2RR to CH_3OH on the $\text{Ti}_3\text{C}_2-3^*\text{OH}-4^*\text{F}$ surface model. If only the most stable intermediate states are considered (dark blue lines) the limiting span is $^*\text{CH}_2\text{OH} \rightarrow \text{CH}_3\text{OH}_{(\text{aq})}$. However, when the less-stable $^*\text{HOCH}_2$ intermediate state is incorporated into the analysis, the step $^*\text{HOCH}_2 \rightarrow \text{CH}_3\text{OH}_{(\text{aq})}$ is not energetically demanding and the limiting span switches to $^*\text{CO}_2 \rightarrow ^*\text{COH}$ with a decrease in the $G_{\max}(U)$. A change in the limiting span $^*\text{CH}_2\text{OH} \rightarrow \text{CH}_3\text{OH}_{(\text{aq})}$ is also observed for the other surfaces with high $^*\text{F}$ coverage as shown in Fig. S14–S16 in Section S8 in the ESI.[†]

So far, we have addressed the underestimation of the selectivity toward CH_3OH for surfaces with high $^*\text{F}$ content. From our results, we conclude that for large $^*\text{F}$ composition, the selectivity trend is in the order of $\text{CH}_3\text{OH} > \text{CH}_4 > \text{H}_2 > \text{CO} > \text{HCOOH}$, which is still not entirely consistent with the experiments where CO is the major product followed by CH_3OH .²⁸ In the following section, we discuss the importance of coadsorbates in the search for less-stable configurations that direct the selectivity toward CO . However, before we move on to the next section, there is still one thing that deserves further explanation. The reader may wonder why we just claimed that the selectivity toward CH_3OH is larger than toward CH_4 when the $G_{\max}(U)$ of CH_3OH and CH_4 are the same (*cf.* Fig. 5d). We come to this conclusion because electrocatalytic processes with a smaller number of proton–electron transfer steps result in a larger exchange current density, which translates into higher electrocatalytic activity.⁶¹ While CH_3OH formation requires 6 proton–electron transfer steps, the formation of CH_4 requires 8, and therefore we argue that CH_3OH formation is kinetically preferred over CH_4 formation, in line with experimental observations.²⁸

5.2 Effect of including coadsorbates

In this section we evaluate the CO formation in the presence of coadsorbates and discuss how this situation can modify the catalytic activity. We have shown in the previous section that our predicted selectivity toward CO disagrees with experiments.²⁸ The main limitation for CO formation is the strong interaction of $^*\text{CO}_2$ and $^*\text{CO}$ with the surface, which limits CO production at low and large cathodic potentials, respectively (Fig. 4c). Therefore, we have evaluated how the presence of coadsorbates may affect the CO formation. To do so, we have studied the CO_2RR to CO on the neighboring available site (see Fig. S18 in Section S11 in the ESI[†]) and evaluated how $^*\text{CO}_2$, $^*\text{COOH}$ and $^*\text{CO}$ coadsorbates affect the CO production.



Fig. 6a depicts the CO desorption energy without coadsorbates and when $^*\text{CO}_2$, $^*\text{COOH}$ and $^*\text{CO}$ are present as coadsorbates, respectively. From Fig. 6a, it is evident that the presence of neighboring species reduces the CO desorption energy for all the surface models. Larger changes are observed for $^*\text{CO}_2$ and $^*\text{COOH}$ compared to $^*\text{CO}$, which we attribute to increased steric effects for the former intermediates. Interestingly, as shown in Fig. 6b, the presence of a $^*\text{CO}_2$ coadsorbate leads to a decrease of the descriptor $G_{\max}(U)$, which we trace to a less-stable configuration compared to the situation where the two intermediates are not direct neighbors. Note that the terminology “surfA” and “surfB” is used to distinguish between directly adjacent sites and non-adjacent sites, respectively. It turns out that the new less-stable configuration ($^*\text{CO}_2$,_{surfA}) reduces the $^*\text{CO}$ desorption energy compared to the situation without the coadsorbate or without direct proximity to CO ($^*\text{CO}_2$,_{surfB}). Detailed free-energy diagrams including also $^*\text{COOH}$ and $^*\text{CO}$ as coadsorbates for all the surfaces can be found in Section S12 of the ESI.†

Finally, we evaluate how the presence of coadsorbates changes the activity descriptor $G_{\max}(U)$ as a function of the

applied potential (Fig. 6c). For the potential dependency of the other surfaces, we direct the reader to Section S12 of the ESI.† Note that in Fig. 6c, we have considered four different scenarios for the $^*\text{CO}$ desorption, namely: (i) no coadsorbate is present or (ii) $^*\text{CO}_2$, (iii) $^*\text{COOH}$, or (iv) $^*\text{CO}$ is present as a coadsorbate. As explained above, when no coadsorbate is present, $^*\text{CO}_2$ adsorption and $^*\text{CO}$ desorption are limiting the span at low and large cathodic potentials (*i.e.*, $^*\text{CO}_2 \rightarrow \text{CO}_{(\text{g})}$ and $^*\text{CO} \rightarrow \text{CO}_{(\text{g})}$, respectively (blue line). As evident from Fig. 6b, the presence of coadsorbates modifies the energetics in such a way that $G_{\max}(U)$ is reduced with the concomitant increase of the catalytic activity. Particularly the presence of a neighboring $^*\text{CO}_2$ substantially reduces the $G_{\max}(U)$ and thus promotes CO formation. Interestingly, at working conditions (*i.e.*, $U = -0.4$ V vs. RHE), it is no longer the $^*\text{CO}$ desorption that is limiting the span but rather formation of the $^*\text{COOH}$ intermediate ($^*\text{CO}_2 \rightarrow ^*\text{COOH}$). As shown by the horizontal lines in Fig. 6c, there is a threshold electrode potential where $^*\text{CO}$ desorption is the limiting step and switching to higher cathodic potentials does not lead to an increase in catalytic activity. Therefore,

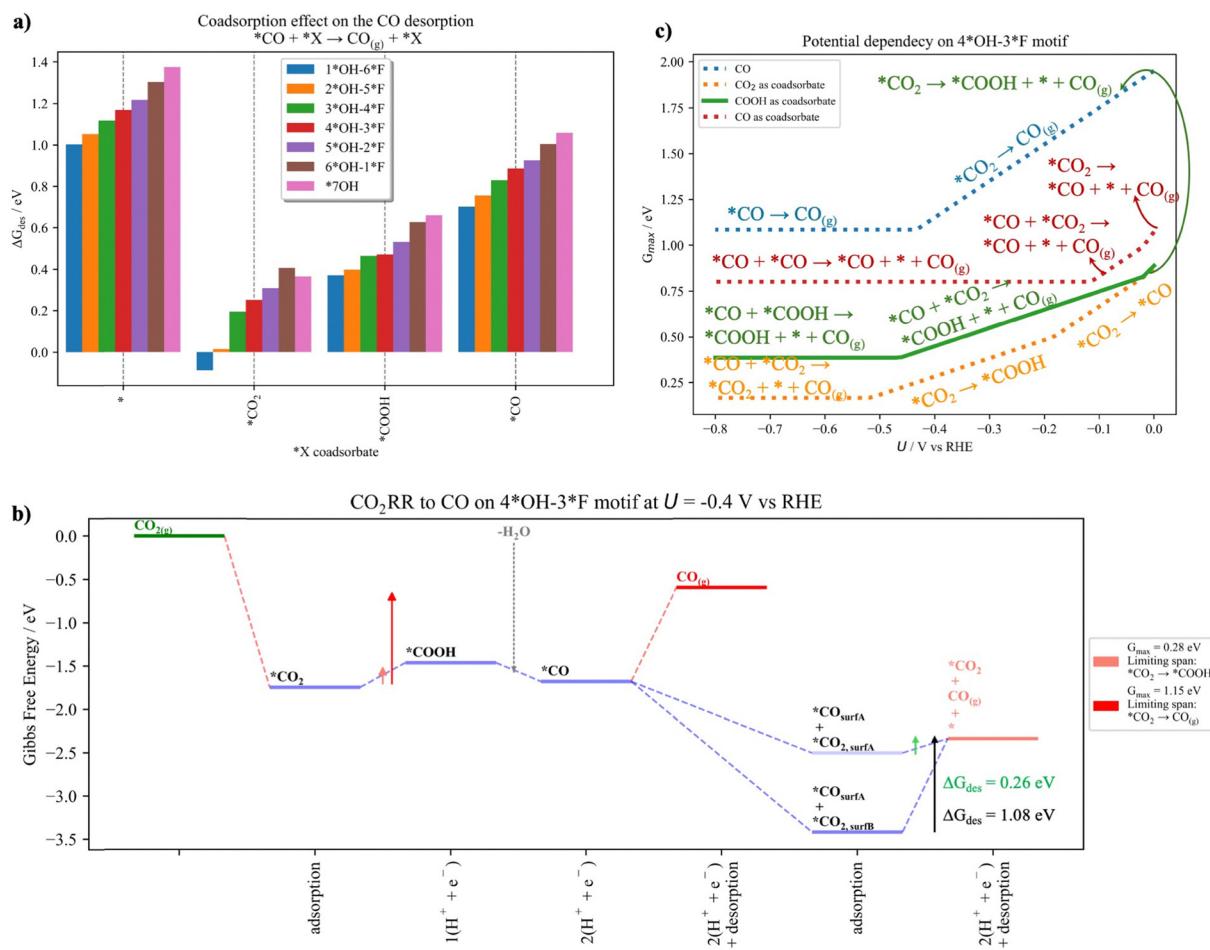


Fig. 6 (a) Effect of coadsorbate species (*i.e.*, $*$, $^*\text{CO}_2$, $^*\text{COOH}$ and $^*\text{CO}$) on the CO desorption for the different studied surfaces models, (b) Gibbs free energy diagram for CO_2RR to CO with the presence of a $^*\text{CO}_2$ coadsorbate (light blue bars) at $U = -0.4$ V vs. RHE for the $\text{Ti}_3\text{C}_2-4^*\text{OH}-3^*\text{F}$ surface. The terminology “surfA” and “surfB” is used to distinguish between directly adjacent sites and non-adjacent sites, respectively. Red and salmon arrows are used to schematically represent the limiting span for $\text{CO}_{(\text{g})}$, while black and green arrows represent the CO desorption energy when including $*$ and $^*\text{CO}_2$ as coadsorbates, respectively. (c) Potential dependence of the CO_2RR to CO with the presence of coadsorbates for the $\text{Ti}_3\text{C}_2-4^*\text{OH}-3^*\text{F}$ surface model.

Table 1 $G_{\max}(U)$ at $U = -0.4$ V vs. RHE of the different products for the different surface models in the CO₂RR and HER over Ti₃C₂T_x MXene. The last column indicates the selectivity order for each surface predicted using the descriptor $G_{\max}(U)$ when less-stable intermediates and coadsorbates are considered

Surface	$G_{\max}(U = -0.4$ V vs. RHE)					Selectivity
	H ₂ (g)	CO(g)	HCOOH _(aq)	CH ₃ OH _(aq)	CH ₄ (g)	
7*OH	0.61	0.70	1.71	0.70	0.68	H ₂ > CH ₄ > CO > CH ₃ OH > CO > HCOOH
6*OH-1*F	0.62	0.48	1.73	0.48	0.48	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH
5*OH-2*F	0.64	0.39	1.73	0.39	0.39	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH
4*OH-3*F	0.64	0.28	1.73	0.28	0.28	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH
3*OH-4*F	0.64	0.25	1.68	0.25	0.25	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH
2*OH-5*F	0.63	0.20	1.66	0.20	0.20	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH
1*OH-6*F	0.62	0.14	1.63	0.14	0.14	CO = CH ₃ OH = CH ₄ > H ₂ > HCOOH

weakening the *CO interaction with the surface by producing less-stable configurations due to the presence of coadsorbates is an efficient way to control CO formation. To conclude, we suggest that after CO₂ adsorbs on the neighboring site, *CO will efficiently desorb and the remaining *CO₂ can be converted into another C₁ species (Fig. 6b).

5.3 Selectivity trends

Finally, let us evaluate the selectivity on the different surfaces if less-stable intermediate states and coadsorbates effects are considered in our analysis. Table 1 summarizes the calculated $G_{\max}(U)$ at working conditions ($U = -0.4$ V vs. RHE) for the different products and the predicted selectivity based on such values. We refer the interested reader to Section S13 of the ESI[†] for the potential dependency of all surface models when less-stable intermediate states and coadsorbates are included in the analysis.

From Table 1, different three clear trends can be observed: (i) while HER is favored for the fully hydroxylated surface, the selectivity shifts toward CO₂RR with increasing *F content, which is in line with experiments.^{27,28} (ii) The larger the *F coverage, the more active (*i.e.*, lower $G_{\max}(U)$) is the Ti₃C₂T_x surface for CO₂RR. (iii) HCOOH formation is not feasible for the selected systems. Focusing on the most selectively generated CO₂RR products, the predicted selectivity using the descriptor $G_{\max}(U)$ is the same for CO, CH₃OH, and CH₄. Nevertheless, reactions with a smaller number of proton-electron transfer steps are generally faster (higher exchange current density) than reactions that require more proton-electron transfer steps. As the CO₂RR to CO, CH₃OH, and CH₄ consists of 2, 6, and 8 proton-electron transfer steps, respectively, we suggest that the selectivity for the *F-covered surfaces is in the order: CO > CH₃OH > CH₄ > H₂ > HCOOH, which is consistent with the experiments.²⁸ This qualitative agreement with the experimental data could only be achieved if less-stable intermediate states and coadsorbates are included in our computational analysis.

6 Conclusions

In the present manuscript, we introduce a new way of thinking for the investigation of proton-coupled electron transfer steps with applications in energy conversion and storage from a

computational perspective. While the traditional approach used in electronic structure theory to derive activity and selectivity trends is based on the construction of free-energy diagrams using the thermodynamically most stable intermediate structures only, we ponder on the importance of incorporating also less-stable intermediate states and coadsorbates in the analysis of adsorption free energies. We benchmark our new concept using the CO₂RR over Ti₃C₂T_x (T_x = *OH and/or *F) surfaces with different *OH/*F coverage as a model system. We demonstrate that less-stable intermediate states lead to energetically favored reaction channels, and only by including these new pathways, we find good agreement with experimental data. Notably, we also show based on rotational barriers that the less-stable intermediates are stable on the electrode surface, which corroborates their feasibility as new reaction channels. In addition, we demonstrate the key aspect of including coadsorbate species in the analysis, which also give rise to less-stable configurations and are particularly relevant for CO formation. We believe that the introduced concept of considering less-stable intermediate states and coadsorbates when analyzing free-energy diagrams to predict activity and selectivity trends is not a unique feature of the CO₂RR over Ti₃C₂T_x, but rather a universal feature to be considered in future computational research to develop improved materials for energy conversion applications.

Data availability

The data supporting this article have been included as part of the ESI.[†]

Conflicts of interest

There are no conflicts to declare.

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References

- 1 F. Barbir, Transition to Renewable Energy Systems with Hydrogen as an Energy Carrier, *Energy*, 2009, **34**, 308–312.
- 2 Q. Lu and F. Jiao, Electrochemical CO₂ Reduction: Electrocatalyst, Reaction Mechanism, and Process Engineering, *Nano Energy*, 2016, **29**, 439–456.
- 3 M. Götz, J. Lefebvre, F. Mörs, A. M. Koch, F. Graf, S. Bajohr, R. Reimert and T. Kolb, Renewable Power-to-Gas: A Technological and Economic Review, *Renewable Energy*, 2016, **85**, 1371–1390.
- 4 A. Mazza, E. Bompard and G. Chicco, Application of Power to Gas Technologies in Emerging Electrical Systems, *Renewable Sustainable Energy Rev.*, 2018, **92**, 794–806.
- 5 P. Lozano-Reis, H. Prats, P. Gamallo, F. Illas and R. Sayós, Multiscale Study of the Mechanism of Catalytic CO₂ Hydrogenation: Role of the Ni(111) Facets, *ACS Catal.*, 2020, **10**, 8077–8089.
- 6 P. Lozano-Reis, H. Prats, R. Sayós and F. Illas, Limitations of Free Energy Diagrams to Predict the Catalytic Activity: The Reverse Water Gas Shift Reaction Catalyzed by Ni/TiC, *J. Catal.*, 2023, **425**, 203–211.
- 7 P. Lozano-Reis, P. Gamallo, R. Sayós and F. Illas, Comprehensive Density Functional and Kinetic Monte Carlo Study of CO₂ Hydrogenation on a Well-Defined Ni/CeO₂ Model Catalyst: Role of Eley-Rideal Reactions, *ACS Catal.*, 2024, **14**, 2284–2299.
- 8 M. A. A. Aziz, A. A. Jalil, S. Triwahyono and A. Ahmad, CO₂ Methanation over Heterogeneous Catalysts: Recent Progress and Future Prospects, *Green Chem.*, 2015, **17**, 2647–2663.
- 9 R. P. Ye, J. Ding, W. Gong, D. M. Argyle, Q. Zhong, Y. Wang, K. C. Rusell, Z. Xu, G. A. Russel, Q. Li, M. Fan and Y. G. Yao, CO₂ hydrogenation to high-value products via heterogeneous catalysis, *Nat. Commun.*, 2019, **10**, 5698.
- 10 D. Dolz, R. De Armas, P. Lozano-Reis, A. Morales-García, F. Viñes, R. Sayós and F. Illas, Understanding the Reverse Water Gas Shift Reaction over Mo₂C MXene Catalyst: A Holistic Computational Analysis, *ChemCatChem*, 2024, **16**, e202400122.
- 11 P. Li, Z. Qi and D. Yan, Rare Earth Er-Nd Dual Single-AtomicCatalysts for Efficient Visible-light Induced CO₂ Reduction to C_nH_{2n+1}OH (N = 1, 2), *Angew. Chem., Int. Ed.*, 2024, **63**, e202411000.
- 12 P. Chen, B. Lei, X. Dong, H. Wang, J. Sheng, W. Cui, J. Li, Y. Sun, Z. Wang and F. Dong, Rare-Earth Single-Atom La-N Charge-Transfer Bridge on Carbon Nitride for Highly Efficient and Selective Photocatalytic CO₂ Reduction, *ACS Nano*, 2020, **14**, 15841–15852.
- 13 S. Ji, Y. Qu, T. Wang, Y. Chen, G. Wang, X. Li, J. Dong, Q. Y. Chen, W. Zhang, Z. Zhang, S. Liang, R. Yu, Y. Wang, D. Wang and Y. Li, Rare-Earth Single Erbium Atoms for Enhanced Photocatalytic CO₂ Reduction, *Angew. Chem., Int. Ed.*, 2020, **59**, 10651–10657.
- 14 X. Li, L. Li, G. Chen, X. Chu, X. Liu, C. Naisa, D. Pohl, M. Löffler and X. Feng, Accessing parity-forbidden *d*-*d* transitions for photocatalytic CO₂ reduction driven by infrared light, *Nat. Commun.*, 2023, **14**, 4034.
- 15 S. Nitopi, E. Bertheussen, B. S. Scott, X. Liu, K. A. Engstfeld, S. Horch, B. Seger, E. L. I. Stephens, K. Chan, C. Hahn, J. K. Norskov, T. F. Jaramillo and I. Chorkendorff, Progress and Perspectives of Electrochemical CO₂ Reduction on Copper in Aqueous Electrolyte, *Chem. Rev.*, 2019, **119**, 7610–7672.
- 16 L. E. Clark, C. Hahn, T. F. Jaramillo and A. T. Bell, Electrochemical CO₂ Reduction over Compressively Strained CuAg Surface Alloys with Enhanced Multi-Carbon Oxygenate Selectivity, *J. Am. Chem. Soc.*, 2017, **139**(44), 15848–15857.
- 17 W. Fan, Y. Liu, C. Zhang, X. Chen, D. He, M. Li, Q. Hu, X. Jiao, Q. Chen and Y. Xie, Confined CO in a Sandwich Structure Promotes C-C Coupling in Electrocatalytic CO₂ Reduction, *Mater. Horiz.*, 2024, **11**, 4183–4189.
- 18 L. R. L. Ting, O. Piqué, S. Y. Lim, M. Tanhaei, F. Calle-Vallejo and B. S. Yeo, Enhancing CO₂ Electroreduction to Ethanol on Copper-Silver Composites by Opening an Alternative Catalytic Pathway, *ACS Catal.*, 2020, **10**, 4059–4069.
- 19 A. D. Handoko, F. Wei, Jenndy, B. S. Yeo and Z. W. Seh, Understanding Heterogeneous Electrocatalytic Carbon Dioxide Reduction Through Operando Techniques, *Nat. Catal.*, 2018, **1**, 922.
- 20 Q. Kong, X. An, Q. Liu, L. Xie, J. Zhang, Q. Li, W. Yao, A. Yu, Y. Jiao and C. Sun, Copper-based Catalysts for the Electrochemical Reduction of Carbon Dioxide: Progress and Future Prospects, *Mater. Horiz.*, 2023, **10**, 698–721.
- 21 J. Peng, Z. Shi, J. Jiang, P. Zhang, J. P. Hsu and N. Li, Charge-orbital Synergistic Engineering of TM@Ti₃C₂O_{1-x}B_x for Highly Selective CO₂ Electrochemical Reduction, *Mater. Horiz.*, 2023, **10**, 4278–4292.
- 22 Y. Y. Birdja, E. Pérez-Gallent, M. C. Figueiredo, A. J. Göttle, F. Calle-Vallejo and M. T. M. Koper, Advances and Challenges in Understanding the Electrocatalytic Conversion of Carbon Dioxide to Fuels, *Nat. Energy*, 2019, **4**, 732–745.
- 23 X. Zhang, S. X. Guo, K. A. Gandionco, A. M. Bond and J. Zhang, Electrocatalytic Carbon Dioxide Reduction: from Fundamental Principles to Catalyst Design, *Mater. Today Adv.*, 2020, **7**, 100074.
- 24 P. Saha, S. Amanullah and A. Dey, Selectivity in Electrochemical CO₂ Reduction, *Acc. Chem. Res.*, 2022, **55**, 134–144.
- 25 O. Christensen, A. Bagger and J. Rossmeisl, The Missing Link for Electrochemical CO₂ Reduction: Classification of CO vs. HCOOH Selectivity via PCA, Reaction Pathways and Coverage Analysis, *ACS Catal.*, 2024, **14**, 2151–2161.
- 26 B. Anasori, M. R. Lukatskaya and Y. Gogotsi, 2D metal carbides and nitrides (MXenes) for energy storage, *Nat. Rev. Mater.*, 2017, **2**, 16098.
- 27 H. Bao, Y. Qui, X. Peng, J. Wang, Y. Mi, S. Zhao, X. Liu, Y. Liu, R. Cao, L. Zhuo, J. Ren, J. Sun, J. Luo and X. Sun, Isolated Copper Single Sites for High-performance Electro-reduction of Carbon Monoxide to Multicarbon products, *Nat. Commun.*, 2021, **12**, 283.
- 28 S. Krishnan, S. Marimuthu, M. K. Singh and D. K. Rai, Two-dimensional Ti₃C₂T_x MXene Nanosheets for CO₂



Electroreduction in Aqueous Electrolytes, *Energy Adv.*, 2023, **2**, 1166.

29 J. Greely, T. F. Jaramillo, J. Bonde, I. Chorkendorff and J. K. Norskov, Computational High-throughput Screening of Electrocatalytic Materials for Hydrogen Evolution, *Nat. Mater.*, 2006, **5**, 909.

30 B. C. Yeo, H. Nam, H. Nam, M. C. Kim, H. W. Lee, S. C. Kim, S. O. Won, D. Kim, K. Y. Lee, S. Y. Lee and S. S. Han, High-throughput Computational-Experimental Screening protocol for the Discovery of Bimetallic Catalysts, *npj Comput. Mater.*, 2021, **7**, 137.

31 S. Wang, V. Petzold, V. Tripkovic, J. Kleis, J. G. Howalt, E. Skúlason, E. M. Fernández, G. Jones, A. Toftlund, H. Falsig, M. Björketun, F. Studt, F. Abild-Pedersen, J. Rossmeisl, J. K. Norskov and T. Bligaard, Universal Transition State Scaling Relations for (De)Hydrogenation over Transition Metals, *Phys. Chem. Chem. Phys.*, 2011, **13**, 20760–20765.

32 S. Wang, B. Temel, J. Shen, G. Jones, L. C. Grabow, F. Studt, T. Bligaard, F. Abild-Pedersen, C. H. Christensen and J. K. Norskov, Universal Brønsted-Evans-Polanyi Relations for C–C, C–O, C–N, N–O, N–N and O–O Dissociation Reactions, *Catal. Lett.*, 2011, **141**, 370–373.

33 H. Ooka, J. Huang and K. S. Exner, The Sabatier Principle in Electrocatalysis: Basics, Limitations, and Extensions, *Front. Energy Res.*, 2021, **9**, 654460.

34 Z. Chen, M. R. Gao, N. Duan, J. Zhang, Y. Q. Zhang, T. Fan, J. Zhang, Y. Dong, J. Li, Q. Liu, X. Yi and J. L. Luo, Tuning Adsorption Strength of CO₂ and its Intermediates on Tin Oxide-based Electrocatalyst for Efficient CO₂ Reduction towards Carbonaceous Products, *Appl. Catal., B*, 2020, **277**, 119252.

35 O. Piqué, F. Viñes, F. Illas and F. Calle-Vallejo, Elucidating the Structure of Ethanol-Producing Active Sites at Oxide-Derived Cu Electrocatalysts, *ACS Catal.*, 2020, **10**, 10488–10494.

36 Y. Xie, P. Ou, X. Wang, Z. Xu, Y. C. Li, Z. Wang, J. E. Huang, J. Wicks, C. McCallum, N. Wang, Y. Wang, T. Chen, B. T. W. Lo, D. Sinton, J. C. Yu, Y. Wang and E. H. Sargent, High Carbon Utilization in CO₂ Reduction to Multicarbon Products in Acidic Media, *Nat. Catal.*, 2022, **5**, 564–570.

37 D. Bao, Q. Zhang, F. L. Meng, H. X. Zhong, M. M. Shi, Y. Zhang, J. M. Yan, Q. Jiang and X. B. Zhang, Electrochemical Reduction of N₂ under Ambient Conditions for Artificial N₂ Fixation and Renewable Energy Storage Using N₂/NH₃ Cycle, *Adv. Mater.*, 2017, **29**, 1604799.

38 X. Yu, Y. Xu, L. Li, M. Zhang, W. Qin, F. Che and M. Zhong, Coverage Enhancement Accelerates Acidic CO₂ Electrolysis at Ampere-level Current with High Energy and Carbon Efficiency, *Nat. Commun.*, 2024, **15**, 1711.

39 K. S. Exner, Controlling Stability and Selectivity in the Competing Chlorine and Oxygen Evolution Reaction over Transition Metal Oxide Electrodes, *ChemElectroChem*, 2019, **6**, 3401–3409.

40 K. S. Exner, Designing Criteria for the Competing Chlorine and Oxygen Evolution Reactions: Avoid the OCl Adsorbate to Enhance Chlorine Selectivity, *Phys. Chem. Chem. Phys.*, 2020, **22**, 22451.

41 G. Kresse and J. Hafner, Ab Initio Molecular Dynamics for Liquid Metals, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **47**, 558–561.

42 G. Kresse and J. Furthmüller, Efficient Iterative Schemes for Ab Initio Total-energy Calculations Using a Plane-wave Basis Set, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.

43 G. Kresse and J. Furthmüller, Efficiency of Ab-initio Total Energy Calculations for Metals and Semiconductors Using a Plane-wave Basis Set, *Comput. Mater. Sci.*, 1996, **6**, 15–50.

44 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.

45 S. Grimme, J. Antony, S. Ehrlich and H. Krieg, A Consistent and Accurate Ab Initio Parametrization of Density Functional Dispersion Correction (DFT-D) for the 94 Elements H–Pu, *J. Chem. Phys.*, 2010, **132**, 154104.

46 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jónsson, Origin of the Overpotential for Oxygen Reduction at a Fuel-Cell Cathode, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.

47 J. A. Garrido Torres, P. C. Jennings, M. H. Hansen, J. R. Boes and T. Bligaard, Low-Scaling Algorithm for Nudged Elastic Band Calculations Using Surrogate Machine Learning Model, *Phys. Rev. Lett.*, 2019, **122**, 156001–156005.

48 M. López, K. S. Exner, F. Viñes and F. Illas, Computational Pourbaix Diagrams for MXenes: A Key Ingredient toward Proper Theoretical Electrocatalytic Studies, *Adv. Theory Simul.*, 2023, **6**, 2200217.

49 L. Meng, L.-K. Yan, F. Viñes and F. Illas, Effect of Terminations on the Hydrogen Evolution Reaction Mechanism on Ti₃C₂ MXene, *J. Mater. Chem. A*, 2023, **11**, 6886–6900.

50 K. S. Exner, A universal Descriptor for the Screening of Electrode Materials for Multiple-Electron Processes: Beyond the Thermodynamic Overpotential, *ACS Catal.*, 2020, **10**, 12607–12617.

51 S. Razzaq and K. S. Exner, Materials Screening by the Descriptor G_{max}(η): The Free-Energy Span Model in Electrocatalysis, *ACS Catal.*, 2023, **13**, 1740–1758.

52 K. S. Exner, On the Mechanistic Complexity of Oxygen Evolution: Potential-dependent Switching of the Mechanism at the Volcano Apex, *Mater. Horiz.*, 2023, **10**, 2086–2095.

53 L. Granda-Marulanda, A. Rendón-Calle, S. Builes, F. Illas, M. T. M. Koper and F. Calle-Vallejo, A Semiempirical Method to Detect and Correct DFT-Based Gas-Phase Errors and Its Application in Electrocatalysis, *ACS Catal.*, 2020, **10**, 6900–6907.

54 R. Urrego-Ortiz, S. Builes, F. Illas and F. Calle-Vallejo, Gas-phase Errors in Computational Electrocatalysis: a Review, *EES Catal.*, 2024, **2**, 157–179.

55 E. Tayyebi and K. S. Exner, Refining Free-Energy Calculations for Electrochemical Reactions: Unveiling Corrections beyond Gas-Phase Errors for Solvated Species and Ions, *J. Phys. Chem. C*, 2024, **128**, 13732–13742.



56 D. Zhang, Z. Wang, F. Liu, P. Yi, L. Peng, Y. Chen, L. Wei and H. Li, Unraveling the pH-Dependent Oxygen Reduction Performance on Single-Atom Catalysts: From Single-to-Duella-Sabatier Optima, *J. Am. Chem. Soc.*, 2024, **146**, 3210–3219.

57 D. Zhang, F. She, J. Chen, L. Wei and H. Li, Why Do Weak-Binding M-N-C Single-Atom Catalysts Posses Anomalously High Oxygen Reduction Activity?, *J. Am. Chem. Soc.*, 2025, **157**, 6076–6086.

58 M. J. Janik, C. D. Taylor and M. Neurock, First-Principles Analysis of the Initial Electroreduction Steps of Oxygen over Pt(111), *J. Electrochem. Soc.*, 2009, **156**, B126–B135.

59 V. Tripkovic, E. Skúlason, S. Siahrostmai, J. K. Nørskov and J. Rossmeisl, The Oxygen Reduction Reaction Mechanism on Pt(111) from Density Functional Theory Calculations, *Electrochim. Acta*, 2010, **55**, 7975–7981.

60 T. Cheng, H. Xiao and W. A. Goddard, Free-Energy Barriers and Reaction Mechanisms for the Electrochemical Reduction of CO on the Cu(100) Surface, Including Multiple Layers of Explicit Solvent at pH 0, *J. Phys. Chem. Lett.*, 2015, **23**, 4767–4773.

61 K. S. Exner, I. Sohrabnejad-Eskan and H. Over, A Universal Approach to Determine the Free Energy Diagram of an Electrocatalytic Reaction, *ACS Catal.*, 2018, **8**, 1864–1879.

