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# Deciphering the carrier dynamics enhancement in WO<sub>3</sub>-containing composites: an ultrafast transient reflectance investigation†

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In recent years, the quest for WO<sub>3</sub>-based composites with expanded optical absorption and elevated photocatalytic and photoelectrochemical performance has intensified. Despite these pursuits, a clear mechanistic understanding of the efficiency gains has been a missing piece, hindering further optimization. This study uses the transient reflectance (TR) technique to probe the intricacies of carrier dynamics in WO<sub>3</sub>, Li<sub>0.1</sub>WO<sub>3</sub>, and the FeOOH/Li<sub>0.1</sub>WO<sub>3</sub> composite, and found that under near-ultraviolet (NUV) illumination, the FeOOH/Li<sub>0.1</sub>WO<sub>3</sub> system demonstrates a photo-carrier lifetime that is approximately twofold and tenfold longer than that of Li<sub>0.1</sub>WO<sub>3</sub> and WO<sub>3</sub>, respectively. Moreover, under near-infrared (NIR) irradiation, the FeOOH/Li<sub>0.1</sub>WO<sub>3</sub> composite sustains carriers for nearly four times longer than the Li<sub>0.1</sub>WO<sub>3</sub> counterpart. Our in-depth analysis points to the hole-filling effect from lithium intercalation and the FeOOH-mediated passivation of oxygen vacancies as the dual drivers behind the significantly extended carrier lifetimes, which favor processes such as charge separation, transfer, extraction, and storage. By employing femtosecond TR measurements to dissect the photo-carrier behavior in these systems, this study sheds light on the intrinsic mechanisms that have been obscured using traditional approaches. This work aims to contribute pivotal insights, potentially paving the way for the next generation of high-performance WO<sub>3</sub> composites in energy conversion technologies.

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## Introduction

Tungsten oxides (WO<sub>x</sub>, where  $x \leq 3$ ) stand as pioneers among visible light-responsive semiconductors and have long been celebrated for their exceptional attributes, including high electron mobility, remarkable electron transfer and storage capabilities, and commendable electrochemical stability – all of which are coupled with the advantage of being earth-abundant.<sup>1</sup> Their potential shines

brightly in photocatalysis and photoelectrochemistry. However, the unadorned WO<sub>x</sub> struggles with relatively low efficiency, primarily due to the swift recombination of photo-generated electrons (e<sup>-</sup>) and holes (h<sup>+</sup>). Moreover, WO<sub>x</sub>'s optical absorption is confined to the near-ultraviolet (NUV) to visible spectrum,



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which captures only a fraction of the solar spectrum, as over half of the sun's energy lies in the near-infrared (NIR) region.

In recent years, the pursuit of  $\text{WO}_x$ -containing composites with enhanced carrier behaviors and an expanded optical absorption spectrum has seen some remarkable advancements. For example, the typical band gap of  $\text{WO}_x$ , which ranges from 2.6 to 2.8 eV, can be narrowed to approximately 0.6–0.9 eV by embedding cations such as  $\text{H}^+$ ,  $\text{Li}^+$ ,  $\text{Na}^+$ ,  $\text{K}^+$ , and  $\text{Cs}^+$  into the tunneling hexagonal structured  $\text{WO}_3$ , which extends the absorption spectrum from NUV to NIR.<sup>2–4</sup> Furthermore, coupling  $\text{WO}_x$  with other semiconductors like  $\text{C}_3\text{N}_4$ ,  $\text{BiVO}_4$ , and  $\text{FeOOH}$  has been shown to significantly enhance the charge carrier separation efficiency, which in turn boosts electron transfer and storage capacities.<sup>3,5–8</sup> The broadening of the absorption spectrum is attributed to the formation of a new valence band, which is supplied by the partially occupied 5d orbitals of  $\text{W}^{5+}$  ions.<sup>2</sup> Despite these strides, a comprehensive understanding of the mechanisms that enhance carrier dynamics remains obscured, thereby hindering further optimization of material performance.

Traditionally, time-resolved photoluminescence (TRPL) has been the technique of choice for probing photo-carrier dynamics in photocatalytic (PC)/photoelectrochemical (PEC) systems.<sup>9–11</sup> However, TRPL faces limitations when it comes to examining the NIR region, where low photoluminescence quantum yields pose a challenge for detection.<sup>12</sup> Furthermore, the complexity of PC/PEC processes, which involve a cascade of interrelated carrier reactions such as generation, transfer, hole filling, trapping, and recombination, complicates the picture. While TRPL is a powerful tool for studying slower dynamics like carrier trapping and recombination, it may introduce significant uncertainties when measuring ultrafast carrier dynamics on the order of picoseconds, such as hole filling.<sup>10,12</sup>

In this study, we introduce transient reflectance (TR) spectroscopy with femtosecond time resolution to elucidate the carrier dynamics in PC/PEC systems. This technique has allowed us to capture the behavior of photo-carriers in  $\text{Li}_{0.1}\text{WO}_3$  with broadened optical absorption, and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  with superior PC/PEC performance,<sup>8,13,14</sup> using pristine  $\text{WO}_3$  as the control. Our TR findings reveal that under 400 nm illumination, photo-carriers in  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  endure more than twice as long as those in  $\text{Li}_{0.1}\text{WO}_3$  and ten times longer than those in  $\text{WO}_3$ . Under 800 nm illumination, the  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  composite sustains photo-carriers nearly four times longer than the  $\text{Li}_{0.1}\text{WO}_3$  alone. Further analysis indicates that the significantly extended carrier lifetimes in  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , which favor processes such as charge separation, transfer, extraction, and storage, could be attributed to the hole-filling effect from lithium intercalation and the passivation of oxygen vacancy defects by  $\text{FeOOH}$ . By delving into the factors influencing photo-carrier behavior in these sunlight harvesting systems using femtosecond TR measurements, this study aims to offer novel insights into the intrinsic mechanisms that may have been elusive using conventional methods.

## Experiments and results

### Static characterizations

$\text{WO}_3$  and  $\text{Li}_{0.1}\text{WO}_3$  were synthesized *via* an *in situ* solvothermal approach, while the  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  composite was fabricated by depositing  $\text{FeOOH}$  onto the surface of  $\text{Li}_{0.1}\text{WO}_3$  nanorods. Detailed procedures for sample preparation are delineated in the ESI,<sup>†</sup> S1. Conventionally,  $\text{WO}_3$  synthesized using this method adopts an orthorhombic crystal structure. However, the incorporation of lithium through intercalation provides both  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  with a pure hexagonal phase of  $\text{WO}_3$ ,<sup>2,3,15,16</sup> in accordance with the joint committee on powder diffraction standards (JCPDS) reference 33-1387. This phase purity is corroborated by the X-ray diffraction (XRD, ARL Equinox 3500) patterns presented in Fig. 1a. It is noteworthy that, due to the amorphous nature of  $\text{FeOOH}$  in the  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  composite, no distinct diffraction signals attributable to  $\text{FeOOH}$  were detected in the XRD analysis.

The inset of Fig. 1b presents the absorption spectra for the trio of samples, as measured using a UV-vis-NIR spectrophotometer (Lambda 950, PerkinElmer, UK). To elucidate the energy structural properties, a detailed analysis of the absorption spectra was conducted employing the Tauc equation, represented as  $(\alpha h\nu)^n = B(h\nu - E_g)$ ,<sup>17</sup> where  $\alpha$  signifies the absorption coefficient,  $h\nu$  denotes the photon energy,  $B$  is a proportionality constant,  $E_g$  is the band gap energy, and  $n$  equals 1/2 for an indirect-bandgap semiconductor, as depicted in Fig. 1b. The optical data reveal that all samples exhibit robust absorption in the NUV region, characterized by an absorption edge of approximately 2.68 eV, indicated by the green dotted line in Fig. 1b. Furthermore,  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  manifest an additional absorption band in the visible to near-infrared (vis-NIR) region, evidenced by a reduced band edge of approximately 0.78 eV, marked by the red

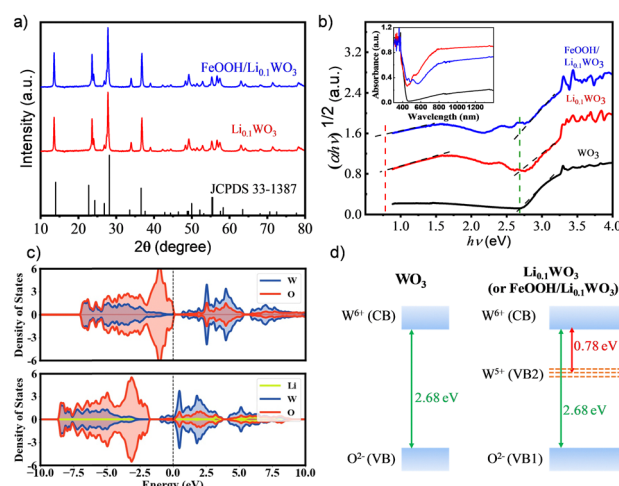


Fig. 1 (a) XRD patterns of  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ ; (b) Tauc plots of the UV-vis-NIR absorption spectra (inset) of  $\text{WO}_3$ ,  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ ; (c) TDOS spectra of  $\text{Li}_{0.1}\text{WO}_3$  as well as  $\text{WO}_3$  for comparison; and (d) schematic diagram of the band structure of  $\text{WO}_3$  and  $\text{Li}_{0.1}\text{WO}_3$  (or  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ ) based on the absorption spectra.

dotted line in Fig. 1b. The former represents the intrinsic band gap energy of  $\text{WO}_3$ , whereas the latter is induced by  $\text{Li}^+$  intercalation, which can be explained by the energy distribution of the density of states (DOS) derived from first-principles calculations. The DOS results depicted in Fig. 1c show that in  $\text{WO}_3$ , the 2p orbitals of  $\text{O}^{2-}$  are fully occupied, and the 5d orbitals of  $\text{W}^{6+}$  are vacant, constituting the valence band (VB, also noted as VB1 for  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  for distinguishing from VB2) and the conduction band (CB), respectively, as illustrated in the left panel of Fig. 1d.<sup>2,18</sup> When lithium is incorporated to form  $\text{Li}_{0.1}\text{WO}_3$ , the Fermi level ( $E_f$ ) shifts upward towards the CB, as compared to the pristine  $\text{WO}_3$ . Moreover, nearly all lithium metal atoms contribute to the upper CB, indicating that they are effectively ionized, thereby providing a substantial supply of free electrons. As a result, the originally vacant 5d orbitals of  $\text{W}^{6+}$  become partially filled, leading to the formation of  $\text{W}^{5+}$ . These partially filled 5d orbitals of  $\text{W}^{5+}$  create a new valence band (VB2) positioned below the CB, which accounts for the visible to near-infrared (vis-NIR) absorption, as shown in the right panel of Fig. 1d. Additionally, as shown in Fig. 1a, the absorption spectra of  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  are strikingly similar, both differing from that of  $\text{FeOOH}$ .<sup>19</sup> This similarity suggests that  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  shares a comparable spectroscopic mechanism with  $\text{Li}_{0.1}\text{WO}_3$ , and the incorporation of  $\text{FeOOH}$  has minimal impact on the primary energy structure of the latter.

The interpretation of the optical characteristics is corroborated by X-ray photoelectron spectroscopy (XPS, Thermo Scientific ESCALAB Xi+) analysis. Fig. S1 in the ESI,<sup>†</sup> S2 displays the survey XPS spectra for the three samples. The spectra reveal not only the ubiquitous presence of W and O elements but also the distinct presence of Li and Fe, confirming the successful incorporation of Li and  $\text{FeOOH}$  into  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , respectively. The Li 1s XPS spectrum for  $\text{Li}_{0.1}\text{WO}_3$ , as shown in Fig. 2a, is deconvoluted into two peaks: a principal peak at 56.3 eV and a shoulder peak at 58.1 eV. The latter is indicative of the interaction between Li and the lattice oxygen in  $\text{Li}_{0.1}\text{WO}_3$ , while the former is attributed to  $\text{Li}^+$  ions within the hexagonal tunnels of  $\text{WO}_3$ ,<sup>20,21</sup> consistent with the DOS calculations suggesting complete ionization of lithium atoms in  $\text{Li}_{0.1}\text{WO}_3$ .

Additionally, the induced presence of  $\text{W}^{5+}$  ions, a consequence of  $\text{Li}^+$  intercalation, is detectable. Fig. 2b–d illustrate that the W 4f spectra for all three samples can be decomposed into two doublets: one for  $\text{W}^{6+}$  at 35.5 and 37.6 eV, and another for  $\text{W}^{5+}$  at 34.3 and 36.4 eV, respectively, indicating the presence of  $\text{W}^{5+}$  ions in each sample. In  $\text{WO}_3$ , the  $\text{W}^{5+}$  ion peak area ratio of 3.15% is likely due to oxygen vacancy (OV) defects.<sup>22</sup> In  $\text{Li}_{0.1}\text{WO}_3$ , this ratio increases almost twofold to 5.26%, which can be attributed to lithium intercalation and is further explained as follows: In the hexagonal  $\text{Li}_{0.1}\text{WO}_3$  crystal structure, with uniformly dispersed trigonal and hexagonal tunnels, the intracrystalline hexagonal tunnels exhibit a high affinity for  $\text{Li}^+$  ions, which can be represented by the equation  $\text{WO}_3 + \text{Li}^+ + \text{e}^- \rightleftharpoons [\text{WO}_3^- \text{Li}^+]$ .<sup>2,3,15</sup> This process allows for the introduction of a significant number of  $\text{Li}^+$  ions, leading to the

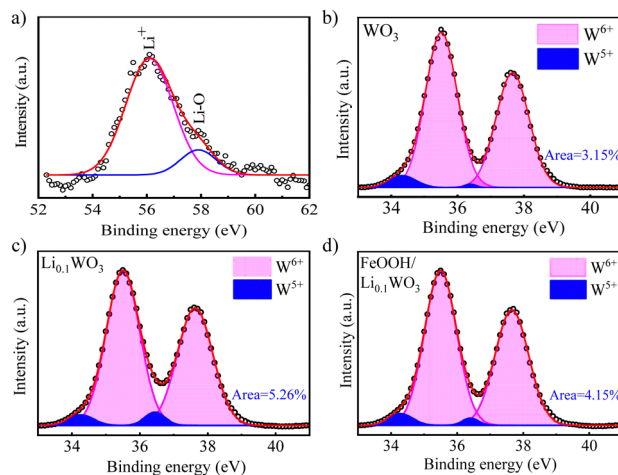


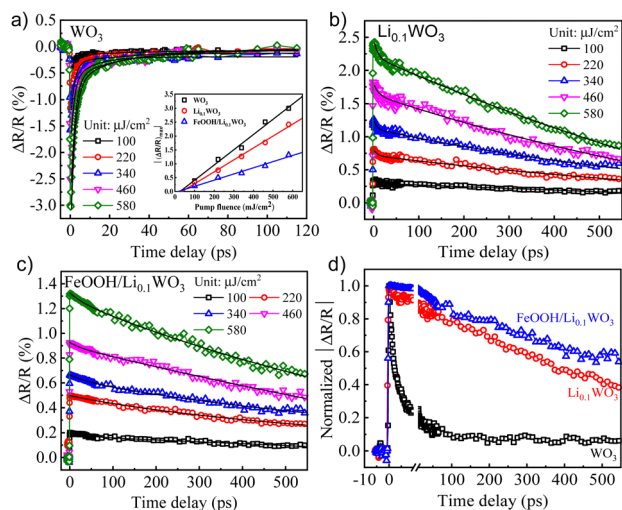
Fig. 2 (a) Li 1s XPS spectrum of  $\text{Li}_{0.1}\text{WO}_3$  and W 4f XPS spectra of (b)  $\text{WO}_3$ , (c)  $\text{Li}_{0.1}\text{WO}_3$  and (d)  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ .

reduction of a substantial amount of  $\text{W}^{6+}$  to  $\text{W}^{5+}$  to maintain the charge balance, thereby increasing the  $\text{W}^{5+}$  ratio in  $\text{Li}_{0.1}\text{WO}_3$  compared to  $\text{WO}_3$ . When  $\text{Li}_{0.1}\text{WO}_3$  is combined with  $\text{FeOOH}$ , the  $\text{W}^{5+}$  ratio decreases to 4.15%. Prior research has indicated that the amorphous  $\text{FeOOH}$  layer may partially passivate OVs in the material.<sup>23</sup> The reduction in the  $\text{W}^{5+}$  ratio in  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  is likely due to the depopulation of OVs passivated by  $\text{FeOOH}$ .

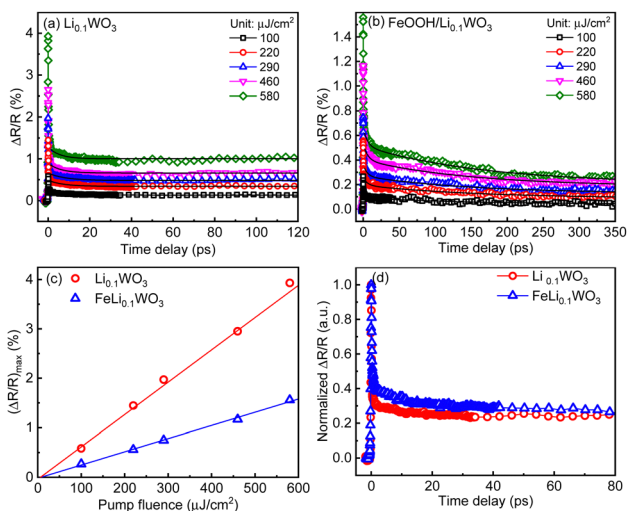
### Ultrafast transient reflectance (TR) measurements and data analysis

As previously outlined, the loading of  $\text{FeOOH}$  has a negligible effect on the energy structure of  $\text{Li}_{0.1}\text{WO}_3$ . Therefore, the dynamic spectral features of  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , detailed in the subsequent discussion, are primarily ascribed to the photo-carriers originating from  $\text{Li}_{0.1}\text{WO}_3$  rather than  $\text{FeOOH}$ . In the TR measurements, two distinct pump wavelengths were employed: a 400 nm wavelength to provide insights into the photo-carrier dynamics across all samples under visible light irradiation, and an 800 nm wavelength to explore those of  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  under NIR irradiation. Meanwhile, an 800 nm wavelength was consistently utilized as the probe regardless of which pump is used. Additional specifics of the TR experimental setup are elaborated in the ESI,<sup>†</sup> S3.

Fig. 3 and 4 display the TR measurement results for the 400 nm and 800 nm excitations, respectively. Following 400 nm excitation, Fig. 3 indicates that the three samples respond differently to the 800 nm probe. Specifically,  $\text{WO}_3$  shows a decrease in reflectance at 800 nm, while  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  exhibit an opposite trend. This variation in response is likely due to the distinct band structure distributions of the materials, as further discussed in the ESI,<sup>†</sup> S4. It is recommended that, while different samples may exhibit varied responses to the 800 nm probe, the probe should consistently capture the dynamics of photo-carriers within the CBs of all samples.



**Fig. 3** (a)–(c) The pump fluence dependent TR kinetics for  $\text{WO}_3$ ,  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  under 400 nm illumination, respectively. (d) The normalized TR traces for three samples at a pump fluence of  $340 \mu\text{J cm}^{-2}$ . Inset of (a): plots of  $(\Delta R_t/R)_{\text{max}}$  as a function of pump fluence. Black solid curves in (a)–(c) are fitting curves using eqn (1) for the extraction of the depopulation rate constants  $k_1$  and  $k_2$ .



**Fig. 4** (a) and (b) The pump fluence dependent TR kinetics for  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  under 800 nm illumination, respectively; (c) plots of  $(\Delta R_t/R)_{\text{max}}$  as a function of pump fluence; (d) normalized TR traces for two samples at a pump fluence of  $220 \mu\text{J cm}^{-2}$ . Black solid curves on (a) and (b) are fitting curves using eqn (1) for the extraction of the depopulation rate constants  $k_1$  and  $k_1$ .

The dynamics triggered by 400 nm excitation primarily involve two processes: a rapid change in reflectivity to the maximum  $(\Delta R_t/R)_{\text{max}}$  within the first picosecond, and a slower recovery process that extends over tens to hundreds of picoseconds. The former is typically assigned to the photo-induced  $e^-/h^+$  assemblage on the CB/VB right after the photon illumination, while the latter should be linked to the relaxation of these photo-carriers through mechanisms like defect trapping and electron–hole recombination.<sup>24</sup> Obviously, photo-carriers in

$\text{WO}_3$  relax much more quickly than those in  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , as illustrated in Fig. 3d. For the 800 nm excitation, except for the above two processes, an additional quick recovery process of less than 2 ps is observed. Considering that the photon energy of 800 nm (1.55 eV) is much larger than the band gap (0.78 eV) traversed, the initially excited carriers should carry much excess energy. The above quick recovery should be the dissipation process of the excess energy through phonon radiation.<sup>25</sup> These carrier assemblage and excess energy dissipation processes are normal and have been well-documented in many semiconductors and insulators,<sup>26,27</sup> thus they will not be further discussed here. In the following, we aim to interpret the complex slow recovery process on a time scale to uncover the underlying mechanisms that boost the superior PC/PEC performances of these materials.

As is known to all, when materials are illuminated by photons with energy exceeding the band gap, electrons will be excited from the VB to assemble on the CB. The concentration of these excited electrons assembled on the CB, denoted as  $n_0$ , can be determined using the formula  $n_0 = Fa/hv$ , where  $F$  represents the incident photon fluence,  $\alpha$  is the material's absorption coefficient at the pump wavelength, and  $h\nu$  corresponds to the pump photon energy. As illustrated in the inset of Fig. 3a and c, under both 400 nm and 800 nm excitation, the maximum relative change in reflectivity,  $|\Delta R_t/R|_{\text{max}}$ , scales linearly with the pump fluence  $F$  (or equivalently,  $n_0$ ) for all samples which leads to the connection as  $|\Delta R_t/R|_{\text{max}} = bn_0$  with a linear coefficient  $b$ . We further hypothesize that the transient reflectivity change  $\Delta R_t/R$  and the transient electron density  $n_t$  are governed by the same recovery function  $f(t)$ , such that  $\Delta R_t/R = |\Delta R_t/R|_{\text{max}}f(t) = bn_0f(t) = bn_t$ . Therefore, the observed slow recovery is indicative of the electron depopulation process within the probed volume.

In semiconductors, the depopulation photo-carriers is typically described by the rate equation  $-dn_t/dt = k_1n_t + k_2n_t^2 + k_3n_t^3$ ,<sup>25,28</sup> where the terms on the right-hand side correspond to: (1) first-order trap state-mediated recombination; (2) second-order non-geminate or free carrier recombination; and (3) three-body Auger recombination.<sup>29</sup> Here,  $k_1$ ,  $k_2$  and  $k_3$  are the respective depopulation rate constants. Given the low pump fluence ( $100\text{--}580 \mu\text{J cm}^{-2}$ ) used in this study, Auger recombination can be disregarded. Consequently, the rate equation simplifies to  $-dn_t/dt = k_1n_t + k_2n_t^2$ , whose analytic solution is given by

$$n_t = k_1n_0 / [(k_2n_0 + k_1)e^{k_1t} - k_2n_0]. \quad (1)$$

By fitting the recovery processes of the three samples with eqn (1), as depicted by the solid black curves in Fig. 3a–c and 4a and b, the corresponding depopulation rate constants  $k_1$  and  $k_2$  can be extracted. For all the samples, the fluctuation of the extracted constants across different pump fluences is minimal; thus, an average value from all fluences ranging from 100 to  $580 \mu\text{J cm}^{-2}$  is used for the discussion, as listed in Table 1. By calculating  $k_2 \times n_0$  (Table 1) and comparing it with  $k_1$ , we found that the former is consistently two or three orders of magnitude

**Table 1**  $k_1$  and  $k_2$  for  $\text{WO}_3$ ,  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  under 400 nm and 800 nm excitation.  $n_0$  is calculated using  $n_0 = F\lambda/h\nu$  with  $F = 330 \mu\text{J cm}^{-2}$  (an average value of the used five pump fluences) and the measured  $\alpha = 2.0 \times 10^4 \text{ cm}^{-1}$ ,  $2.0 \times 10^4 \text{ cm}^{-1}$  and  $1.8 \times 10^4 \text{ cm}^{-1}$  at 400 nm for  $\text{WO}_3$ ,  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , and  $\alpha = 2.2 \times 10^4 \text{ cm}^{-1}$  and  $1.2 \times 10^4 \text{ cm}^{-1}$  at 800 nm for  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , respectively

Pump wavelength	Sample	$k_1$ ( $\times 10^9 \text{ s}^{-1}$ )	$k_2$ ( $\times 10^{-12} \text{ cm}^3 \text{ s}^{-1}$ )	$k_2 \times n_0$ ( $\times 10^7 \text{ s}^{-1}$ )
400 nm	$\text{WO}_3$	4.00	6.32	8.40
	$\text{Li}_{0.1}\text{WO}_3$	0.65	0.88	1.17
	$\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$	0.36	0.39	0.47
800 nm	$\text{Li}_{0.1}\text{WO}_3$	9.75	38.3	99.0
	$\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$	2.16	9.00	13.0

smaller than the later. According to eqn (1), this suggests that the contribution of the second-order mechanisms is significantly weaker. In other words, the electron depopulation processes are predominantly governed by the first-order trap state-mediated recombination. This conclusion is further supported by a more detailed analysis provided in the ESI,<sup>†</sup> S5. Consequently, the subsequent discussion will primarily focus on the first-order mechanism, as characterized by  $k_1$ , for further analysis.

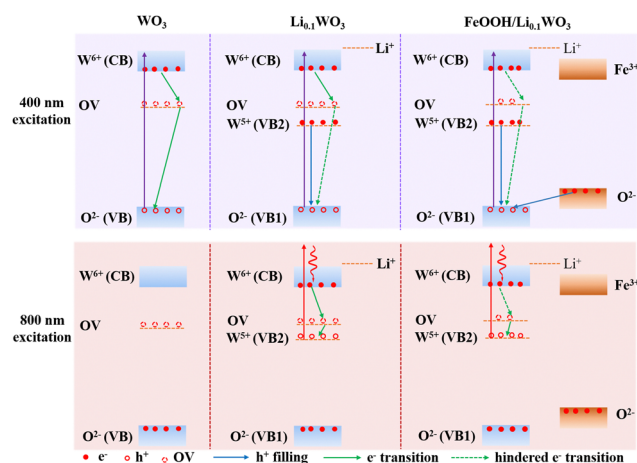
As detailed in Table 1, the depopulation rate constant  $k_1$  exhibits a pronounced sample-dependent behavior under both 400 nm and 800 nm excitation. Specifically, under 400 nm excitation, the value of  $k_1$  for  $\text{WO}_3$  is observed to be  $4.00 \times 10^9 \text{ s}^{-1}$ , which diminishes to  $0.65 \times 10^9 \text{ s}^{-1}$  for  $\text{Li}_{0.1}\text{WO}_3$  and further to  $0.36 \times 10^9 \text{ s}^{-1}$  for  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , reflecting a reduction by factors of 6 and 10, respectively. Meanwhile, under 800 nm excitation,  $k_1$  for  $\text{Li}_{0.1}\text{WO}_3$  is approximately  $9.75 \times 10^9 \text{ s}^{-1}$ , which declines to approximately  $2.16 \times 10^9 \text{ s}^{-1}$  for  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , indicating a reduction by a factor of 4. According to eqn (1), a lower  $k_1$  signifies a slower electron depopulation rate, which in turn implies a longer electron lifetime, and *vice versa*. Consequently, the dynamic measurements suggest that the electron lifetime in  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  is extended compared to that in  $\text{Li}_{0.1}\text{WO}_3$ , and both are substantially longer than the lifetime observed in pure  $\text{WO}_3$ .

## Discussion

Notably, TR measurements indicate that under 400 nm excitation, the electron lifetime in the  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  composite is approximately twice that in  $\text{Li}_{0.1}\text{WO}_3$  and nearly tenfold that in  $\text{WO}_3$ . Meanwhile, under 800 nm excitation, the electron lifetime in  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  is roughly four times longer than in  $\text{Li}_{0.1}\text{WO}_3$ , as detailed in Table 1. We now explore the potential mechanisms that contribute to the extended carrier lifetimes observed in these samples. Firstly, we consider the behavior of charge carriers under 400 nm excitation. Typically,  $\text{WO}_3$  possesses a band gap of 2.68 eV, characterized by the filled 2p orbitals of  $\text{O}^{2-}$  constituting the VB and the unoccupied 5d orbitals of  $\text{W}^{6+}$  forming the CB, as depicted in Fig. 1d and Scheme 1. Concurrently, as previously noted, the synthesis of  $\text{WO}_3$  invariably introduces OVs, which create trap states, illustrated as OV in Scheme 1. Prior density functional theory (DFT) calculations have indicated that in  $\text{WO}_3$ , minor intrinsic OV

defects are more likely to be dispersed rather than aggregated. Furthermore, the trap states arising from these isolated OVs are anticipated to be situated 0.48 to 0.62 eV below the conduction band edge,<sup>30</sup> which aligns with the prevalent perspective that OVs in  $\text{WO}_3$  act as shallow electron donors.<sup>31,32</sup> Upon exposure to 400 nm (3.10 eV) photons, electrons from the VB of  $\text{WO}_3$ , as well as those in  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ , are excited to the CB, leaving behind holes on the VB. Given the indirect bandgap nature of  $\text{WO}_3$ , the direct relaxation of these excited electrons back to the VB is challenging.<sup>29</sup> Consequently, these electrons are typically captured by OVs temporarily before returning to the VB to recombine with holes, as indicated by the green arrows in the  $\text{WO}_3$  panel of Scheme 1 for 400 nm excitation. It is widely understood that the initial trapping process is of the first order, whereas the recombination process is of the second order. The TR data for  $\text{WO}_3$  confirm that the recovery processes are predominantly governed by the first-order mechanism, suggesting that these are indeed the trapping events of electrons by OVs.

When considering  $\text{Li}_{0.1}\text{WO}_3$ , as described above, the intercalation of lithium atoms leads to the reduction of some  $\text{W}^{6+}$  ions to  $\text{W}^{5+}$ . Consequently, in addition to the intrinsic CB and VB (denoted as VB1 in  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$ ) of  $\text{WO}_3$ , the partially occupied 5d orbitals of  $\text{W}^{5+}$  constitute a new valence band, referred to as VB2 in Scheme 1. This results in a reduced band gap of approximately 0.78 eV.<sup>2</sup> Considering that lithium intercalation exerts negligible influence on the intrinsic band gap of  $\text{WO}_3$ , it is reasonable to expect that the positioning of OV states in  $\text{Li}_{0.1}\text{WO}_3$  would closely mirror that in  $\text{WO}_3$ . Accordingly, as illustrated in Scheme 1, VB2 is anticipated to be positioned marginally lower than the OV states. For  $\text{Li}_{0.1}\text{WO}_3$  right after the 400 nm illumination, there should be abundant excited  $e^-/h^+$  on CB/VB1. However, electrons occupying VB2 are likely to swiftly transfer to VB1 to neutralize  $h^+$  (sketched as the blue arrow in Scheme 1 on the  $\text{Li}_{0.1}\text{WO}_3$  panel of 400 nm excitation). This hole filling behavior, which has been observed intensively in semiconductors,<sup>24</sup> leads to a reduction in  $h^+$



**Scheme 1** Schematic diagram of hindered photo-carrier recombination in  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{FeOOH}/\text{Li}_{0.1}\text{WO}_3$  under 400 nm and 800 nm irradiation.

concentration on VB1, impeding the migration of electrons trapped at OV to VB1 ((sketched as the green dotted arrow in Scheme 1 on the  $\text{Li}_{0.1}\text{WO}_3$  panel of 400 nm excitation)). Consequently, the transition of electrons from the CB to OV is obstructed, and  $e^-$  remain in the CB for an extended duration, elucidating the observed prevalence of prolonged first-order relaxation processes in  $\text{Li}_{0.1}\text{WO}_3$  as compared to  $\text{WO}_3$ .

Upon the deposition of FeOOH onto  $\text{Li}_{0.1}\text{WO}_3$ , while the energy level structure remains largely unaltered, OVs in  $\text{Li}_{0.1}\text{WO}_3$  are partially passivated as evidenced by the aforementioned XPS data analysis. Consequently, the trapping effect of OVs on  $e^-$  is diminished. Meanwhile, FeOOH possesses an intrinsic band gap of approximately 2.2 eV, which is marginally smaller than that of  $\text{WO}_3$ .<sup>33</sup> When integrated with  $\text{Li}_{0.1}\text{WO}_3$ , the VB of FeOOH, which is also derived from the 2p orbitals of  $\text{O}^{2-}$ , is anticipated to be marginally higher in energy than that of  $\text{Li}_{0.1}\text{WO}_3$ , as depicted in Scheme 1. In this case, for FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  illuminated by 400 nm photons, in addition to the hole filling from VB2 to VB1 (vertical blue arrow in Scheme 1 on the FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  panel of 400 nm excitation), holes on the VB1 are also neutralized by electrons from the 2p orbitals of  $\text{O}^{2-}$  in FeOOH (oblique blue arrow in Scheme 1 on the FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  panel of 400 nm excitation). The combined effect of the reduced OV trapping and the enhanced hole filling is expected to decelerate electron relaxation, thereby rationalizing the extended carrier lifetime observed in the FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  composite compared to  $\text{Li}_{0.1}\text{WO}_3$  alone.

Secondly, we discuss the behavior of charge carriers under 800 nm excitation. Given the weak absorbance of  $\text{WO}_3$  in the NIR region, no effective TR data were obtained for this material. Therefore, our analysis in this section will focus solely on  $\text{Li}_{0.1}\text{WO}_3$  and FeOOH/ $\text{Li}_{0.1}\text{WO}_3$ . As previously mentioned,  $\text{Li}_{0.1}\text{WO}_3$  and FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  exhibit two distinct bandgaps, 2.68 eV and 0.78 eV, respectively. Upon 800 nm irradiation, the photon energy of 1.55 eV, being lower than the intrinsic band gap of 2.68 eV, will only excite electrons from VB2, leading to the formation of hot electrons and holes on the CB and VB2. Once cooled to the bottom of the CB, electrons are initially captured by OVs before recombining with holes in VB2 (as depicted by the green arrows in Scheme 1 for the 800 nm excitation of both  $\text{Li}_{0.1}\text{WO}_3$  and FeOOH/ $\text{Li}_{0.1}\text{WO}_3$ ). As previously discussed, the OVs in FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  may be partially passivated by the FeOOH component. Therefore, in FeOOH/ $\text{Li}_{0.1}\text{WO}_3$ , the trapping of electrons is obstructed, resulting in an extended residence time for electrons in the CB relative to  $\text{Li}_{0.1}\text{WO}_3$ .

The prolonged electron lifetime should have a positive effect to the PC/PEC efficiencies, which can be interpreted with an example of the photocatalytic process of  $\text{CH}_4$  and  $\text{H}_2\text{O}_2$  reacting to produce  $\text{CH}_3\text{OH}$ . This process can be succinctly summarized as follows: Firstly, when the incident light energy  $h\nu$  is not less than the band gap width  $E_g$ , electrons in the valence band absorb the light energy and transition to the CB, while simultaneously creating holes in the VB. Secondly, the generated  $e^-$  and  $h^+$  migrate to the semiconductor surface under the influence of the electric field or diffusion; finally, the electrons with

reducing power react with  $\text{H}_2\text{O}_2$  to form hydroxyl radicals, which then react with  $\text{CH}_4$  to produce  $\text{CH}_3\text{OH}$ . In this process, to achieve a higher catalytic efficiency, it is necessary to ensure that the photo-generated electrons effectively separate from the photo-generated holes without recombination and successfully migrate to the semiconductor surface. In this study, the TR technique directly observes the lifetime of photo-generated electrons. For example, TR experiments observed a longer lifetime of photo-generated electrons in FeOOH/ $\text{Li}_{0.1}\text{WO}_3$ , which on one hand proves that they are effectively separated from the photo-generated holes without recombination, and on the other hand, the longer survival time also provides ample time for the photo-generated electrons to transfer to the semiconductor surface. This is in accordance with the greatly improved photocatalytic efficiency of FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  compared with  $\text{Li}_{0.1}\text{WO}_3$  and  $\text{WO}_3$  observed previously.<sup>3</sup>

Except for the elevated photocatalytic efficiency,  $\text{WO}_3$ -based composites have been proved to demonstrate significant enhancements in PEC properties. For instance,  $\text{A}_x\text{WO}_3$  (where A represents  $\text{H}^+$ ,  $\text{Na}^+$ ,  $\text{K}^+$ ) not only broadens the optical absorption spectrum to the NIR region but also displays notably enhanced photoelectric anticorrosion capabilities;<sup>4</sup> in addition, under simulated sunlight irradiation, the heterostructured  $\text{WO}_3/\text{FeOOH}$  photoanode has been observed to deliver a photocurrent density nearly double that of pristine  $\text{WO}_3$ .<sup>8</sup> Given the importance of prolonged carrier lifetimes in boosting charge transfer, separation, extraction, and storage efficiencies within sunlight harvesting systems, we propose that the superior PEC performances of these composites may also be partially attributed to the presence of long-lived photo-carriers in them.

## Conclusions

In this study, the TR technique was employed to scrutinize the dynamics of photo-carriers within  $\text{WO}_3$ -containing composites, specifically  $\text{Li}_{0.1}\text{WO}_3$  and FeOOH/ $\text{Li}_{0.1}\text{WO}_3$ , in comparison to pure  $\text{WO}_3$ . The experimental findings indicate that under 400 nm excitation, the photo-carriers in FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  exhibit a significantly extended lifetime, enduring approximately twofold and tenfold longer than those in  $\text{Li}_{0.1}\text{WO}_3$  and pure  $\text{WO}_3$ , respectively. Furthermore, under 800 nm excitation, the photo-carriers in FeOOH/ $\text{Li}_{0.1}\text{WO}_3$  demonstrate a lifetime nearly four times longer than those in  $\text{Li}_{0.1}\text{WO}_3$ . These prolonged lifetimes are posited to enhance the PC/PEC performance, which can be ascribed to the hole filling effect arising from lithium intercalation and the passivation mechanism of OV defects by FeOOH. This investigation not only offers valuable insights for optimizing the PC/PEC performance of  $\text{WO}_3$ -based systems but also introduces innovative perspectives on the underlying mechanisms that may elude traditional investigative approaches.

## Author contributions

Investigation, data analysis and writing – original draft: H. Z.; DFT calculations: T. J.; XPS measurement and data analysis:

M. Z.; sample preparation: H. L.; resources, project administration, and supervision: F. L. All authors discussed the results and contributed to the final manuscript.

## Data availability

The data that support the findings of this study are available from the corresponding author, H. Z., upon reasonable request.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 Y. Yao, D. Sang, L. Zou, Q. Wang and C. Liu, *Nanomaterials*, 2021, **11**, 2136.
- 2 X. Wu, Y. Li, G. Zhang, H. Chen, J. Li, K. Wang, Y. Pan, Y. Zhao, Y. Sun and Y. Xie, *J. Am. Chem. Soc.*, 2019, **141**, 5267–5274.
- 3 Y. Zeng, X. Luo, F. Li, A. Huang, H. Wu, G. Q. Xu and S. L. Wang, *Environ. Sci. Technol.*, 2021, **55**, 7711–7720.
- 4 Z.-X. Liu, D. Zhou, J.-J. Tian and J.-K. Liu, *Adv. Eng. Mater.*, 2023, **26**, 2301003.
- 5 Z.-F. Huang, J. Song, X. Wang, L. Pan, K. Li, X. Zhang, L. Wang and J.-J. Zou, *Nano Energy*, 2017, **40**, 308–316.
- 6 K. Pingmuang, J. Chen, W. Kangwansupamonkon, G. G. Wallace, S. Phanichphant and A. Nattestad, *Sci. Rep.*, 2017, **7**, 8929.
- 7 Z. Ni, Q. Wang, Y. Guo, H. Liu and Q. Zhang, *Catalysts*, 2023, **13**, 579.
- 8 G. Zheng, S. Jiang, M. Cai, F. Zhang and H. Yu, *J. Alloy. Compd.*, 2024, **981**, 173637.
- 9 Y. Li, Y.-K. Peng, L. Hu, J. Zheng, D. Prabhakaran, S. Wu, T. J. Puchtler, M. Li, K.-Y. Wong, R. A. Taylor and S. C. E. Tsang, *Nat. Commun.*, 2019, **10**, 4421.
- 10 D. Grinberg and Y. Paz, *Curr. Opin. Green Sustainable*, 2023, **41**, 100768.
- 11 X. Wang, M. Sayed, O. Ruzimuradov, J. Zhang, Y. Fan, X. Li, X. Bai and J. Low, *Appl. Mater. Today*, 2022, **29**, 101609.
- 12 E. Mitchell, A. Law and R. Godin, *Chem. Commun.*, 2021, **57**, 1550–1567.
- 13 J. Wei and W. Shen, *Colloid. Surface. A*, 2022, **643**, 128754.
- 14 J. Wang, L. Jiang, F. Liu, M. Jia, M. Liu, J. Li and Y. Lai, *Chem. Eng. J.*, 2021, **407**, 127195.
- 15 W. Sun, M. T. Yeung, A. T. Lech, C.-W. Lin, C. Lee, T. Li, X. Duan, J. Zhou and R. B. Kaner, *Nano Lett.*, 2015, **15**, 4834–4838.
- 16 L. Chen, S. Lam, Q. Zeng, R. Amal and A. Yu, *J. Phys. Chem. C*, 2012, **116**, 11722–11727.
- 17 S. Chen, T. Liu, M. Chen, M. Ishaq, R. Tang, Z. Zheng, Z. Su, X. Li, X. Qiao, P. Fan and G. Liang, *Nano Energy*, 2022, **99**, 107417.
- 18 I. Grigioni, M. Abdellah, A. Corti, M. V. Dozzi, L. Hammarström and E. Selli, *J. Am. Chem. Soc.*, 2018, **140**, 14042–14045.
- 19 A. Ramzannezhad, P. Gill and A. Bahari, *BioNanoMaterials*, 2017, **18**, 20170008.
- 20 W. Chen, Q. Xu, Y. S. Hu, L. Q. Mai and Q. Y. Zhu, *J. Mater. Chem.*, 2002, **12**, 1926–1929.
- 21 A. C. Kozen, A. J. Pearse, C.-F. Lin, M. A. Schroeder, M. Noked, S. B. Lee and G. W. Rubloff, *J. Phys. Chem. C*, 2014, **118**, 27749–27753.
- 22 F. Wang, C. Di Valentin and G. Pacchioni, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 073103.
- 23 X. Lu, K.-h. Ye, S. Zhang, J. Zhang, J. Yang, Y. Huang and H. Ji, *Chem. Eng. J.*, 2022, **428**, 131027.
- 24 P. Piatkowski, B. Cohen, F. Javier Ramos, M. Di Nunzio, M. K. Nazeeruddin, M. Grätzel, S. Ahmad and A. Douhal, *Phys. Chem. Chem. Phys.*, 2015, **17**, 14674–14684.
- 25 J. S. Manser and P. V. Kamat, *Nat. Photonics*, 2014, **8**, 737–743.
- 26 Z. Chi, H. Chen, Z. Chen, Q. Zhao, H. Chen and Y. X. Weng, *ACS Nano*, 2018, **12**, 8961–8969.
- 27 C. C. S. Chan, K. Fan, H. Wang, Z. Huang, D. Novko, K. Yan, J. Xu, W. C. H. Choy, I. Lončarić and K. S. Wong, *Adv. Energy Mater.*, 2021, **11**, 2003071.
- 28 R. Qian, H. Zong, J. Schneider, G. Zhou, T. Zhao, Y. Li, J. Yang, D. W. Bahnemann and J. H. Pan, *Catal. Today*, 2019, **335**, 78–90.
- 29 P. J. Boruah, R. R. Khanikar and H. Bailung, *Plasma Chem. Plasma Process.*, 2020, **40**, 1019–1036.
- 30 M. Sachs, J. S. Park, E. Pastor, A. Kafizas, A. A. Wilson, L. Francas, S. Gul, M. Ling, C. Blackman, J. Yano, A. Walsh and J. R. Durrant, *Chem. Sci.*, 2019, **10**, 5667–5677.
- 31 W. Wang, A. Janotti and C. G. Van de Walle, *J. Mater. Chem. C*, 2016, **4**, 6641–6648.
- 32 K. Kato, Y. Uemura, K. Asakura and A. Yamakata, *J. Phys. Chem. C*, 2022, **126**, 9257–9263.
- 33 F. M. Ali, M. Hmadeh, P. G. O'Brien, D. D. Perovic and G. A. Ozin, *ChemNanoMat*, 2016, **2**, 1047–1054.