



Cite this: *J. Mater. Chem. C*,
2024, 12, 2194

Received 12th October 2023,
Accepted 3rd January 2024

DOI: 10.1039/d3tc03720g

rsc.li/materials-c

Oxazine: an anchoring group serving as functional kernels to construct single-molecule switches†

Shi Li,^a Yuxuan Jiang,^{ab} Yudi Wang,^a Dongying Lin,^a Haoyang Pan,^{ac} Yongfeng Wang,^a Stefano Sanvitto^d and Shimin Hou^a   

As a central research target in molecular electronics, molecular switches have been extensively explored over the past decades. We theoretically demonstrate that when linking appropriate conjugated molecules to carbon electrodes, the de/rehydrogenation of 1,4-oxazine linkers efficiently switches single-molecule junctions between a low-conducting and a high-conducting state. This change is attributed to the modified energy gap of the central molecule as well as the charge rearrangement at the molecule–electrode interfaces. Based on the above findings, a single-molecule junction, employing an intramolecular proton transfer reaction as the switching mechanism, has been proposed. This realizes a maximum ON/OFF current ratio as high as 1.5×10^3 . Furthermore, we show that an electrostatic gate field can control the proton transfer process and thus allow specific conductance states to be selected. Our findings provide a new perspective for the design of single-molecule switches relying on the anchoring groups rather than on the specific molecular backbones.

Introduction

Molecular electronics seeks to make full use of the characteristics of molecules to design and construct functional devices with the long-term goal of providing an alternative to traditional silicon-based technologies.^{1–4} As one of the key building blocks of electronic circuits, molecular switches interconverting between a high-conducting (“ON”) and a low-conducting (“OFF”) state have attracted much attention over the years.^{5–7} Two core issues in this field are how to construct single-molecule switches and how to efficiently control their electronic transport characteristics. To this end, various molecules with bistable and multistable states addressable *via* external stimuli, such as an electric field,^{8–15} light,^{16–24} pH,^{25–31} or mechanical forces,^{32–36} have been intensively investigated. For example, as typical photochromic molecules, azobenzene derivatives are able to isomerize reversibly between the *cis* and *trans* conformations under light irradiation and have been incorporated into single-molecule junctions as photoswitching units, realizing significant tuning of single-molecule conductance between the “ON” and “OFF”

states.^{37–41} To the best of our knowledge, most of the studies to date have focused on the central molecule of a junction, while little research has been conducted to explore the potential of the anchoring groups. These link the central molecules to the electrodes and they may possibly work as switching kernels.

The 1,4-oxazine molecule and its derivatives (Chart 1a), like phenoxazines and dioxazines, have sparked much interest owing to their numerous applications in the fields of pharmacology,^{42–45} photophysics^{46–50} and material science.^{51–55} Notably, the H atom bonded to the N atom in an oxazine ring can be controllably removed and re-added under certain conditions, introducing impressive flexibility to the molecule’s chemical properties.^{56–59} Herein, we investigate the electronic transport properties of a set of oxazine-linked single-molecule junctions with carbon electrodes by employing the non-equilibrium Green’s function formalism

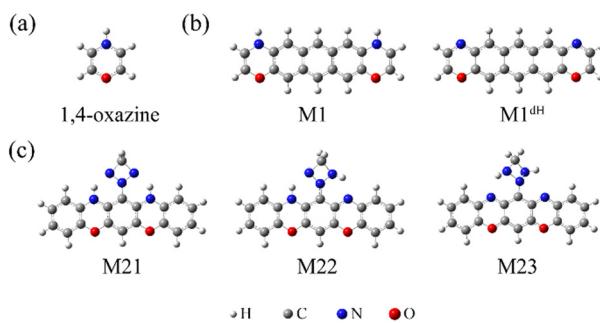


Chart 1 The optimized atomic structures of the isolated 1,4-oxazine (a), M1/M1^{dH} (b), and M21/M22/M23 (c).

^a Key Laboratory for the Physics and Chemistry of Nanodevices, School of Electronics, Peking University, Beijing 100871, China. E-mail: smhou@pku.edu.cn

^b Centre for Nanoscale Science and Technology, Academy for Advanced Interdisciplinary Studies, Peking University, Beijing 100871, China

^c Institute of Spin Science and Technology, South China University of Technology, Guangzhou 511442, China

^d School of Physics, AMBER and CRANN Institute, Trinity College, Dublin 2, Ireland

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3tc03720g>

combined with density functional theory (that is, the so-called NEGF+DFT approach).^{60–64} This choice of carbon electrodes, graphene and single-walled carbon nanotubes (SWCNTs), is motivated by previous experimental and theoretical results demonstrating their advantageous applications in the construction of high-performance single-molecule devices operating at room and even higher temperatures.^{65–70} Our calculations show that, by connecting an anthracene molecule to graphene and SWCNT electrodes, the dehydrogenation of oxazine anchors triggers the switching of the investigated molecular junctions from an “OFF” to an “ON” state along with minor alterations of the junction geometry. This dramatic conductance switching effect is mainly attributed to the reduced energy gap between the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) of the central molecule. Importantly, the conductance switching property of oxazine anchors is robust with respect to the curvature of SWCNT electrodes and moderate structural modifications of the molecular backbone.

Furthermore, based on the above findings, we have considered a molecule with two oxazine rings as well as a triazetidine moiety (Chart 1c), which enables three stable tautomeric forms through intramolecular proton transfer. The molecular junctions incorporating these tautomers exhibit profoundly different low-bias conductance, realizing a maximum ON/OFF current ratio as high as 1.5×10^3 . The intrinsic reaction coordinate (IRC) analysis suggests that the relative stability of the tautomers as well as the energy barrier for their interconversion can be controlled by means of an external electrostatic field, a feature that allows the specific conductance states to be selected.

Computational models and methods

Geometry optimization and electronic structure calculations have been performed by using SIESTA.^{71,72} SIESTA is an efficient DFT package, which implements improved Troullier–Martins pseudopotentials for describing the atomic cores and a finite-range numerical orbital basis set to expand the wave functions of the valence electrons.⁷³ We construct a double zeta plus polarization (DZP) basis set for all the elements including H, C, N, and O. The exchange–correlation functional is treated at the level of the generalized gradient approximation (GGA) within the Perdew–Burke–Ernzerhof (PBE) formulation.⁷⁴ A conjugate gradient scheme is applied for geometry relaxation until the residual forces acting on the atoms are less than $0.02 \text{ eV } \text{\AA}^{-1}$. The vibrational and thermodynamic stabilities of the junctions constructed with SWCNT electrodes are verified by computing the phonon spectrum and performing molecular dynamics simulations (see Sections S1 and S2, ESI[†]).

The electronic transport properties of single-molecule junctions have been investigated by using the SMEAGOL code, which is a practical implementation of the NEGF+DFT approach employing SIESTA as the DFT platform.⁷⁵ The same pseudopotentials, basis set and GGA functional are used for both geometry relaxation and transport. The unit cell of the extended molecule consists of a central anthracene molecule, oxazine anchors, and

two graphene or SWCNT electrodes with an armchair edge passivated with hydrogen atoms. Here, the transport is assumed to be along the z -axis and we always consider periodic boundary conditions in the plane transverse to transport. In order to model two-dimensional graphene electrodes, the distance between the neighboring graphene sheets is set to be 15 \AA . The total transmission function $T(E)$ of the junctions is evaluated as

$$T(E) = \frac{1}{\Omega_{1\text{DBZ}}} \int_{1\text{DBZ}} T(\vec{k}; E) d\vec{k} \quad (1)$$

where $\Omega_{1\text{DBZ}}$ is the area of the one-dimensional Brillouin zone (1DBZ) orthogonal to the transport direction. The k -dependent transmission coefficient $T(\vec{k}, E)$ is obtained as

$$T(\vec{k}, E) = \text{Tr}[\Gamma_L G_M^r \Gamma_R G_M^a] \quad (2)$$

where G_M^r is the retarded Green’s function matrix of the extended molecule and Γ_L (Γ_R) represents the broadening function matrix describing the interaction of the extended molecule with the left-hand (right-hand) side electrode. The 1DBZ is sampled by 200 k -points to obtain smooth transmission curves. In contrast, for molecular junctions with quasi-one-dimensional armchair SWCNT electrodes, a k -point grid of $1 \times 1 \times 1$ is chosen with a large vacuum region surrounding the junctions.

Geometry optimization of the minima and transition states, vibrational frequency and intrinsic reaction coordinate (IRC) calculations have been carried out using the Gaussian16 DFT package.^{76–78} All DFT calculations are performed with the PBE density functional and the 6-311+G(d,p) basis set for H, C, N and O atoms. Harmonic frequency analysis confirms the stationary points as local minima (reactants and products) or first-order saddle points (transition states) and extracts zero-point vibrational energy (ZPE) corrections. IRC calculations are performed to verify that the reactant and the product are connected along the reaction pathway through the transition state.^{79,80}

Results and discussion

First, we construct a graphene-M1-graphene junction in which an anthracene molecule is sandwiched between two semi-infinite armchair-terminated graphene electrodes *via* oxazine anchoring groups. In order to find the equilibrium structure, we systematically vary the separation between the two graphene electrodes and optimize the positions of the anthracene molecule, the two oxazine groups and the atoms at the electrode edges until the total energy reaches a local minimum. The equilibrium transmission spectrum of graphene-M1-graphene (red line) is plotted in Fig. 1a, together with the conducting eigenchannels^{81,82} for selected transmission peaks around the Fermi level (E_F). Clearly, these eigenchannels are not localized on the anthracene molecule but are delocalized along the entire central molecular backbone including the two oxazine anchors. Hence, here we consider M1 (Chart 1b) as the central molecule in the junction rather than anthracene. Combined with the inspection of the frontier molecular orbitals (FMOs) of M1 (see Fig. S6 in ESI[†]), it is confirmed that the transmission peaks centered at -0.50 eV and 1.51 eV originate from the M1 HOMO

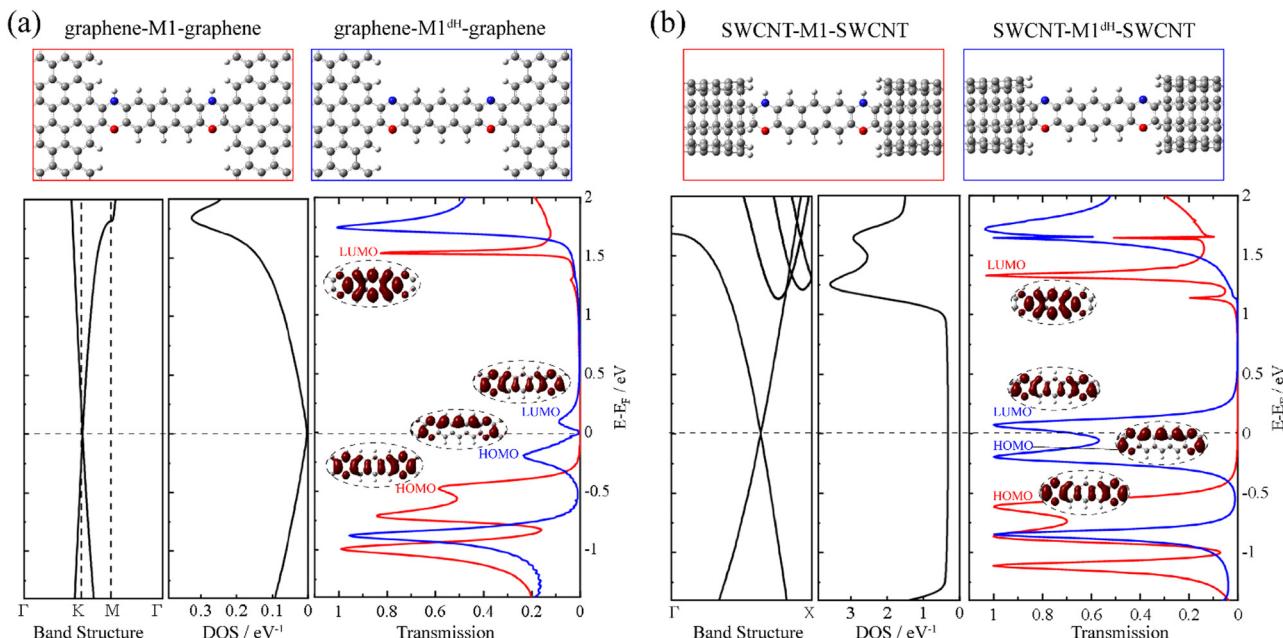


Fig. 1 The optimized atomic structures and the corresponding transmission spectra of graphene-M1-graphene (red line) and graphene-M1^{dH}-graphene (blue line) in (a), and of SWCNT-M1-SWCNT (red line) and SWCNT-M1^{dH}-SWCNT (blue line) in (b), together with the band structure and DOS of graphene in (a) and those of the (5,5) SWCNT in (b). The insets show the conducting eigenchannels for selected transmission peaks around E_F .

and LUMO, respectively. The HOMO-dominated peak decays rapidly toward E_F and the LUMO-dominated peak lies far away from E_F , both making minor contributions to the low-bias conductance of the graphene-M1-graphene junction. Then, we remove the two hydrogen atoms bonded to the nitrogen atoms and re-optimize the atomic structure of the junction, while keeping the electrode-to-electrode separation unchanged. Thus, we obtain a dehydrogenated junction labelled as graphene-M1^{dH}-graphene. It is found that, upon dehydrogenation, the atomic structure of the junction undergoes minor changes; more specifically, all bond lengths vary by no more than 0.03 Å, even those at the molecule-electrode interfaces. A detailed comparison is shown in Fig. S7 (ESI[†]). The equilibrium transmission spectrum of the graphene-M1^{dH}-graphene junction is also plotted in Fig. 1a with a blue line. The vertical axes of these transmission spectra are also set to a logarithmic scale in Fig. S8 (ESI[†]), highlighting their different transmission properties in the vicinity of the Fermi level. The M1^{dH} HOMO- and LUMO-dominated transmission peaks now centered at -0.20 and 0.10 eV, respectively, both lie closer to E_F and would definitely yield a considerable low-bias junction conductance. For example, the conductance of the graphene-M1^{dH}-graphene junction is calculated to be $2.8 \times 10^{-2} G_0$ at a bias voltage of 0.1 V, which is much greater than that ($1.5 \times 10^{-4} G_0$) of the graphene-M1-graphene junction at the same bias. Thus, dehydrogenation of oxazine anchoring groups results in a remarkable conductance enhancement.

It is intriguing to observe that the closer to the Fermi level a state is, the lower the height of the transmission peak in Fig. 1a. This can be ascribed to the gradually decreasing density of states (DOS) of the pristine graphene electrode as the electron

energy approaches the Fermi level.^{70,83} In order to decipher the effect of the electronic structure of graphene on the low transmission phenomena around E_F , we replace the pristine graphene electrodes with (5,5) armchair SWCNT, which is a typical quasi-one-dimensional metal with a constant DOS around E_F . The new molecular junctions are named SWCNT-M1-SWCNT and SWCNT-M1^{dH}-SWCNT and the corresponding transmission spectra are displayed in Fig. 1b. As can be seen, when compared to those of the aforementioned junctions with graphene electrodes, the HOMO- and LUMO-dominated transmission peaks are enormously enhanced in SWCNT-M1-SWCNT and SWCNT-M1^{dH}-SWCNT, while their energy locations undergo limited changes. Consequently, the transmission coefficient for SWCNT-M1-SWCNT is less than 0.01 from -0.3 to 1.0 eV, representing a steady “OFF” state of the switch at low bias voltages. In contrast, the transmission coefficient for SWCNT-M1^{dH}-SWCNT is very high (> 0.60) over a broad energy window around E_F , roughly going from -0.4 to 0.2 eV. This can be used as an “ON” state of the switch.

Another possibility to enhance the low-bias conductance of graphene-M1^{dH}-graphene is to replace the pristine graphene electrodes with nitrogen-doped (N-doped) ones, in which the Dirac point is shifted below E_F and the DOS around E_F increases markedly due to the additional π -electrons brought by the nitrogen dopants.^{70,83,84} Importantly, a remarkable conductance switching effect is also observed in this type of junction. In fact, after substituting one carbon atom with nitrogen in a rectangular unit cell of graphene containing 40 atoms, as shown in Fig. S9 (ESI[†]), the transmission spectrum of the junction after dehydrogenation towers over that of the junction before dehydrogenation in a broad energy range around E_F .

Overall, our calculations demonstrate that the dehydrogenation of oxazine anchoring groups triggers the molecular junctions with different types of carbon electrodes to switch from an “OFF” to an “ON” state along with minor alterations of the junction geometry.

We now move to unveil the origin of this dramatic conductance switching effect. It is well known that the conductance of a single-molecule junction is determined by two fundamental factors: first, the alignment of the FMOs relative to the Fermi level of the electrodes and, second, the electronic coupling between the FMOs and the electrode continuum of states.^{85,86} Coarsely speaking, the former dictates the energy locations of transmission peaks relative to the electrode Fermi level and the latter influences the width and height of the transmission peaks. Evidently, the conductance switching effect along with the dehydrogenation of oxazine anchors is mainly dictated by the first factor. As shown in Fig. 2a, we compare the energy levels of the isolated molecules M1 (red thin lines) and M1^{dH} (blue thin lines), together with the energy centers of the corresponding transmission peaks of the SWCNT-M1-SWCNT (red thick lines) and SWCNT-M1^{dH}-SWCNT (blue thick lines) junctions. It should be noted that the vacuum level is taken as the energy origin for the molecular orbitals of the isolated molecules, whereas in the junctions the Fermi level is set to zero. The M1 HOMO–1, HOMO, and LUMO lie at –3.26, –2.86 and –0.92 eV, respectively, with a HOMO–LUMO gap of 1.94 eV. Upon dehydrogenation, the M1 HOMO becomes unoccupied and the new HOMO and LUMO of the dehydrogenated molecule M1^{dH} (a singlet biradical) lie at –3.22 and –2.85 eV, respectively. That is to say, in the case of isolated molecules, the HOMO–LUMO gap is drastically reduced from 1.94 eV to

0.37 eV due to the dehydrogenation of the oxazine rings. Furthermore, when coupling to the state continuum of the carbon electrodes, the molecular levels further shift by the interaction between the central molecule and the two electrodes. This often occurs with charge transfer at the molecule–electrode interfaces, equalizing the chemical potential across the entire junction. Importantly, this charge-transfer process critically depends on the chemical identity of the anchoring groups. According to the Mulliken population analysis, we find that the central molecule (indicated by the shaded oval in Fig. 2b) is slightly positively charged (+0.03 e) in the SWCNT-M1-SWCNT junction but moderately negatively charged (–0.36 e) in the SWCNT-M1^{dH}-SWCNT one. This suggests that the anchors are turned from electron donors to electron acceptors upon dehydrogenation, leading to a net electron transfer of about 0.4 e from the electrodes to the central molecule. In order to visually comprehend the charge rearrangement induced by dehydrogenation, we investigate the *xy*-plane averaged charge density difference $\delta\rho$ between SWCNT-M1-SWCNT (ρ_0) and SWCNT-M1^{dH}-SWCNT (ρ_{dH}), namely

$$\delta\rho(z) = \rho_{dH}(z) - \rho_0(z) \quad (3)$$

Note that electron transport occurs by convention along the *z*-axis. As shown in Fig. 2b, charge transfer primarily occurs at the electrode–molecule interfaces; the dips in the electrode region and the peaks in the central molecule region (especially the oxazine rings) reflect electron transfer from the electrodes to the central molecule upon dehydrogenation. On the one hand, the enhanced fraction of electrons in the central molecule partially occupies the M1^{dH} LUMO, thus making the LUMO-dominated transmission peak of the SWCNT-M1^{dH}-SWCNT

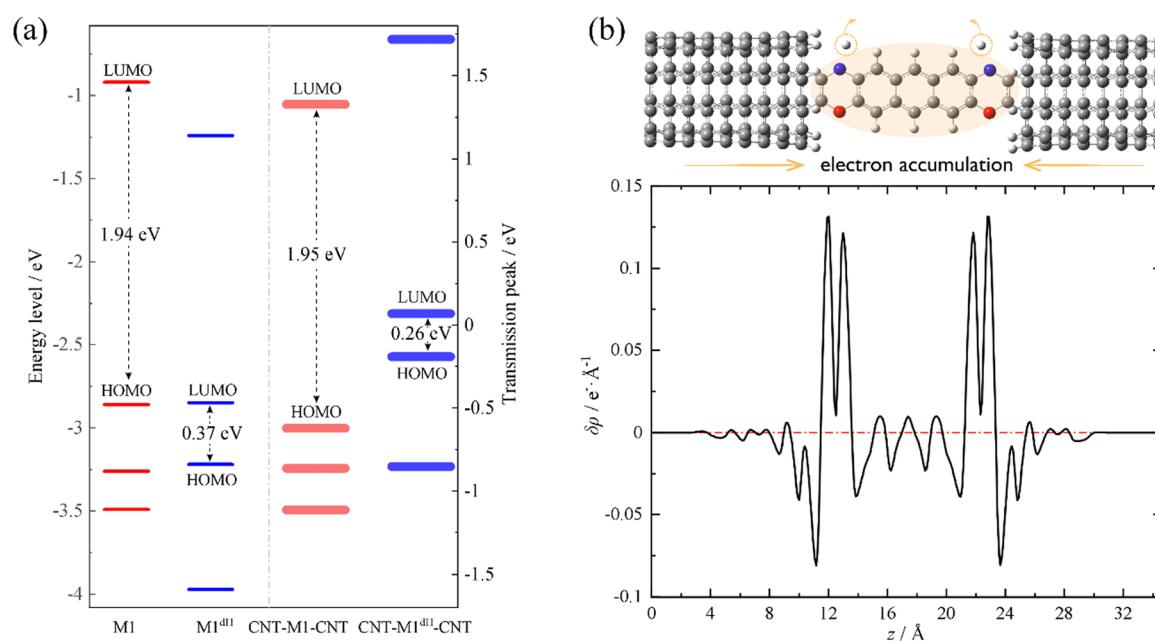


Fig. 2 (a) Molecular energy levels of the isolated M1 (red thin lines) and M1^{dH} (blue thin lines) molecules, and the energy centers of the corresponding transmission peaks in the SWCNT-M1-SWCNT (red thick lines) and SWCNT-M1^{dH}-SWCNT (blue thick lines) junctions. (b) The *xy*-plane averaged charge density difference $\delta\rho(z)$ between the SWCNT-M1-SWCNT and SWCNT-M1^{dH}-SWCNT junctions.

junction hit the Fermi level. On the other hand, the accumulation of electrons promotes electrostatic repulsion in the central molecule and pushes the HOMO. Hence, the energy separation, $\varepsilon_{\text{HOMO-LUMO}}$, between the HOMO-dominated peak (centered at -0.19 eV) and the LUMO-dominated one (centered at 0.07 eV) of SWCNT-M1^{dH}-SWCNT is only 0.26 eV. This gap is narrower than the HOMO-LUMO gap (0.37 eV) of the isolated M1^{dH} molecule. As a comparison, $\varepsilon_{\text{HOMO-LUMO}}$ of SWCNT-M1-SWCNT is 1.95 eV, a value nearly identical to the M1 HOMO-LUMO gap (1.94 eV). All in all, the dehydrogenation of the oxazine rings not only reduces the HOMO-LUMO gap of the isolated molecules but also introduces electron transfer from the electrodes to the central molecule in the junctions, both making the HOMO- and LUMO-dominated transmission peaks approach the Fermi level and hence switching the junction from an “OFF” to an “ON” state.

It should be noted that the conductance switching property of the oxazine anchoring groups is robust, to a certain extent, with respect to variations of both the electrodes and the central molecular backbone. Not only the curvature of the SWCNT electrodes (see Fig. S10a in the ESI[†]) but also the symmetry of the oxazine linkers on both sides of the junction (see Fig. S10b in the ESI[†]) have limited influence on the transport especially around E_F . As shown in Fig. S11 (ESI[†]), the central benzene ring in M1 is substituted with an electron-donating group (oxhydryl) or an electron-withdrawing group (cyano), and the corresponding transport spectra suggest that the switching effect is immune to moderate structural modifications caused by chemical functionalization. Furthermore, when the molecular backbone is altered from an anthracene molecule to a benzene or hexacene (see Fig. S12 in the ESI[†]), the impressive promotion of the low-bias junction conductance upon dehydrogenation has

always been observed although the transport spectra undergo great changes. Previous studies have shown that oxazine derivatives can convert between the two states with and without dehydrogenation under certain experimental conditions.^{56–59} Hence, we believe that 1,4-oxazine has the potential to function as a promising switch unit to construct single-molecule switches by linking suitable conjugated molecules to carbon electrodes.

Based on the above findings, we next propose a molecular switch employing an intramolecular proton transfer reaction mechanism. As sketched in Chart 1c, we consider a molecule (denoted by M23) which has two dehydrogenated oxazine rings at both sides as well as a triazetidine moiety that substitutes on the central benzene. The molecule M23 with a singlet biradical character has two stable tautomeric forms M21 and M22, which can interconvert into each other through intramolecular proton transfer between the oxazine and triazetidine moieties. M21, M22 and M23 are sandwiched between two semi-infinite (5,5) SWCNT electrodes and the electronic transport properties of the corresponding junctions, namely, SWCNT-M21-SWCNT, SWCNT-M22-SWCNT and SWCNT-M23-SWCNT, have been investigated. From Fig. 3b, one can see clearly that the three tautomers deliver extremely different zero-bias transport properties and for each junction there are two prominent transmission peaks in the energy interval $[-0.8$ eV, $+0.8$ eV]. Combined with the inspection of the FMOs of the isolated molecules M21–M23 (see Fig. S13 in the ESI[†]), we confirm that the transmission peaks below and above E_F can be respectively assigned to the HOMO and LUMO of the corresponding central molecule. Importantly, there is a clear trend, namely along with the dehydrogenation of oxazine anchoring groups (in other words, the intramolecular proton transfer from oxazine to triazetidine

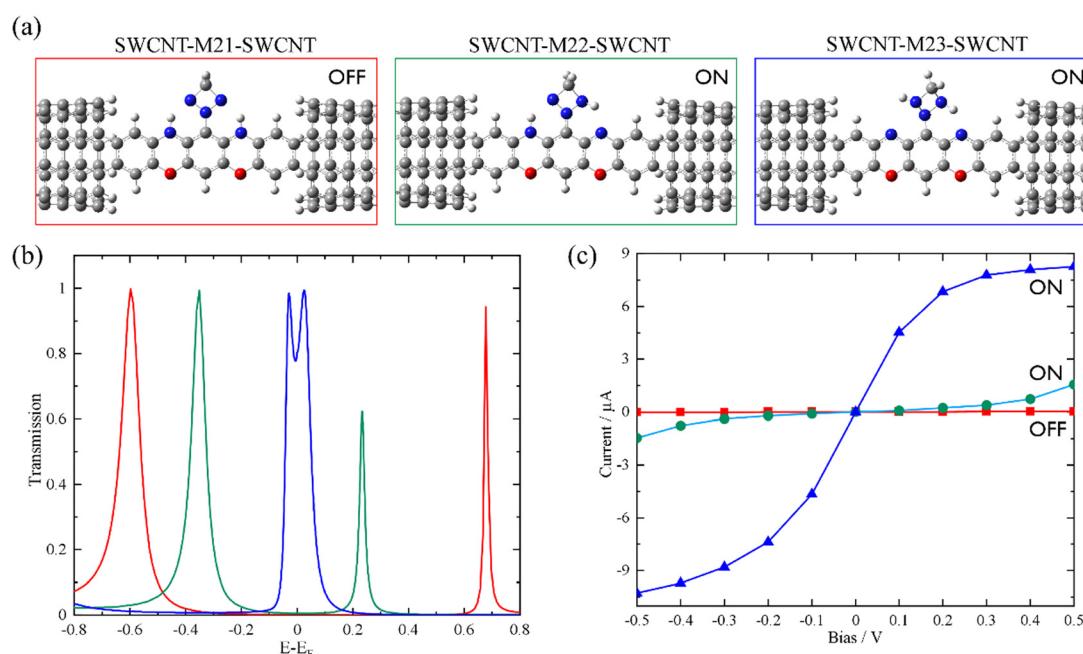


Fig. 3 The optimized atomic structures (a), the equilibrium transmission spectra (b), and the current–voltage characteristics (c) of the SWCNT-M21-SWCNT, SWCNT-M22-SWCNT, and SWCNT-M23-SWCNT junctions.

moieties), the HOMO- and LUMO-dominated transmission peaks gradually approach E_F and result in an increasing transmission coefficient at E_F . More in detail, the HOMO- and LUMO-dominated transmission peaks of SWCNT-M21-SWCNT are located at -0.60 and 0.68 eV, respectively, leading to a transmission coefficient at E_F as small as 9.5×10^{-4} . In contrast, the HOMO- and LUMO-dominated transmission peaks of SWCNT-M22-SWCNT (SWCNT-M23-SWCNT) are centered at -0.35 eV (-0.025 eV) and 0.23 eV (0.025 eV), respectively, leading to an enhanced transmission coefficient of 1.1×10^{-2} (8.2×10^{-1}) at E_F , which is roughly one (three) order(s) of magnitude larger than that of SWCNT-M21-SWCNT. Fig. 3c depicts the current–voltage (I – V) characteristics of these three junctions, which are calculated at finite bias voltages up to ± 0.5 V with an interval of 0.1 V. These I – V curves show that SWCNT-M22-SWCNT and SWCNT-M23-SWCNT are more conducting than SWCNT-M21-SWCNT in the entire applied-voltage range. Note that the electrode-to-electrode separations are held unchanged during the proton transfer process and, in this sense, these three junctions can be regarded as three states of a switch: SWCNT-M21-SWCNT is the “OFF” state, while SWCNT-M22-SWCNT and SWCNT-M23-SWCNT realize two possible “ON” states. The maximum ON/OFF current ratio between SWCNT-M23-SWCNT (SWCNT-M22-SWCNT) and SWCNT-M21-SWCNT is $\sim 1.5 \times 10^3$ (~ 40), found at a bias voltage of 0.1 V (0.4 V). The average values over the voltage range from -0.5 to 0.5 V are approximately 800 and 30 for the SWCNT-M23-SWCNT and SWCNT-M22-SWCNT junctions, respectively.

The HOMO–LUMO gaps of M21, M22, and M23 are respectively calculated to be 1.21 , 0.54 , and 0.07 eV, that is, the intramolecular proton transfer drastically reduces the HOMO–LUMO gap of the isolated molecules, analogous to the variation observed between M1 and M1^{ΔH} (see Fig. S14 in the ESI[†]). According to Mulliken population analysis of these single-molecule junctions, intramolecular proton transfer induces remarkable electron reorganization inside the central molecule, mainly between the oxazine and triazetidine moieties. Specifically, the triazetidine moiety is negatively charged (-0.14 e) in SWCNT-M21-SWCNT but positively charged ($+0.23$ e) in SWCNT-M22 (M23)-SWCNT. However, the charge transfer between the central molecule and the two SWCNT electrodes is small: the central molecule is slightly positively charged ($+0.07$ e) in SWCNT-M21-SWCNT as well as in SWCNT-M22-SWCNT but slightly negatively charged (-0.06 e) in SWCNT-M23-SWCNT. Roughly speaking, although the dehydrogenation converts the oxazine anchoring groups from an electron donor to an electron acceptor, the ensuing charge rearrangement at the molecule–electrode interfaces is moderate, a fact that has a limited influence on the alignment of the FMOs relative to the Fermi level. As a result, the energy separations between the HOMO- and LUMO-dominated transmission peaks of the SWCNT-M21-SWCNT, SWCNT-M22-SWCNT, and SWCNT-M23-SWCNT junctions are 1.28 , 0.58 , and 0.05 eV, respectively, comparable to the HOMO–LUMO gap of the corresponding isolated molecules. Hence, the low-bias conductance enhancement here is mainly attributed to the decreased HOMO–LUMO gaps.

Having established that the designed molecular junction could support three stable conformational forms with robust differences in the low-bias conductance, a possible reversible mechanism among these ON and OFF states is outlined in what follows. This is achieved by controlling the intramolecular proton transfer process *via* an external electrostatic field as in a field-effect transistor configuration. To this end, we have performed the intrinsic reaction coordinate (IRC) calculations of the proton transfer process at the level of isolated molecules. The black lines in Fig. 4 and Fig. S15 (ESI[†]) depict the energy pathways for proton transfer without an external field. It is found that M21 is the most stable structure among these three tautomers, with M22 and M23 respectively lying 0.63 and 1.22 eV above M21. The calculated energy barriers from M21 to M22 and from M22 to M23 are 0.80 and 1.08 eV, respectively. Nevertheless, an external electrostatic field (marked with a black arrow in the inset of Fig. 4) applied in the plane of the molecule and perpendicular to its longitudinal axis could lower the energy barrier height for the proton transfer process. In particular, upon applying an electric field of about 1.1 V \AA^{-1} , the barrier for the M21-to-M22 transformation is reduced to 0.30 eV and now M22 is more stable than M21. An inversion of the relative stability under an external electric field is also obtained for M22 and M23 (see Fig. S15 in the ESI[†]). These calculations suggest that without an external gate field, the single-molecule switch tends to work in the SWCNT-M21-SWCNT form; in other words, at the “OFF” state. However, an external electrostatic gate field could trigger the switch to the “ON” state working in the SWCNT-M22 (M23)-SWCNT form. Although the calculated field strength needed for the stabilization of the “ON” states is rather high for direct experimental realization, appropriate chemical modifications of the central molecule may facilitate a more accessible interconversion at much reduced field strengths. Here, we simply provide a proof of concept that conductance control using an external electrostatic field is feasible for oxazine-anchored molecular junctions.

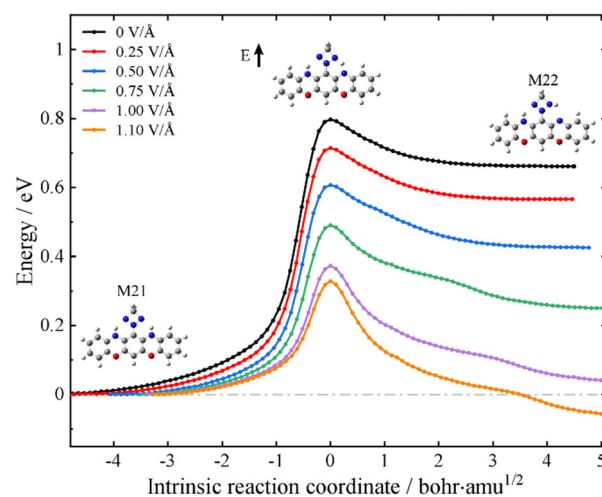


Fig. 4 Energy along the intrinsic reaction path from M21 to M22 in the gas phase. The electric field vector is indicated by a black arrow in the inset.

Conclusions

In summary, by employing the NEGF+DFT approach, we have investigated the electronic transport properties of molecular junctions in which an anthracene molecule is connected to graphene and SWCNT electrodes *via* oxazine anchors. Our calculations show that the dehydrogenation of oxazine anchors triggers the transition of the investigated molecular junctions from an “OFF” to an “ON” state along with minor alterations of the junction geometry. The dehydrogenation of oxazine rings not only reduces the HOMO-LUMO gap of the isolated molecule, but also introduces electron transfer from the electrodes to the central molecule in the junctions. This causes both the HOMO- and LUMO-dominated transmission peaks to approach the Fermi level. Furthermore, a molecule with two oxazine rings and a triazetidine moiety, which enables three stable tautomeric forms through intramolecular proton transfer, has been investigated. By sandwiching it between two (5,5) armchair SWCNT electrodes, these tautomers exhibit profoundly different low-bias conductances and the maximum ON/OFF current ratio is as high as 1.5×10^3 . By applying an external electrostatic field, the relative stability of the tautomers, as well as the energy barrier for their interconversion, can be effectively controlled. Our findings demonstrate that 1,4-oxazine serves as a promising switch kernel when linking appropriate conjugated molecules to carbon electrodes. This provides a new perspective for the design and construction of single-molecule switches relying on the anchoring groups and not on the specific molecular backbones.

Author contributions

Shi Li and Shimin Hou conceived the idea. Shi Li carried out the theoretical calculations, data analysis and wrote the initial draft. Yuxuan Jiang, Yudi Wang, Dongying Lin and Haoyang Pan carried out parts of theoretical calculations and helped to discuss and analyze the calculation results. Yongfeng Wang, Stefano Sanvito and Shimin Hou critically reviewed and helped to revise and polish the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This project was supported by the National Natural Science Foundation of China (Grant No. 21933002 and 22225202) and the High-performance Computing Platform of Peking University. SS thanks the Science Foundation Ireland (AMBER Center grant 12/RC/2278_P2) for financial support.

Notes and references

1 A. Nitzan and M. A. Ratner, *Science*, 2003, **300**, 1384–1389.

- 2 M. L. Perrin, E. Burzuri and H. S. J. van der Zant, *Chem. Soc. Rev.*, 2015, **44**, 902–919.
- 3 F. Evers, R. Kortydar, S. Tewari and J. M. van Ruitenbeek, *Rev. Mod. Phys.*, 2020, **92**, 035001.
- 4 H. Fu, X. Zhu, P. Li, M. Li, L. Yang, C. Jia and X. Guo, *J. Mater. Chem. C*, 2022, **10**, 2375–2389.
- 5 L. Sun, Y. A. Diaz-Fernandez, T. A. Gschneidtner, F. Westerlund, S. Lara-Avila and K. Moth-Poulsen, *Chem. Soc. Rev.*, 2014, **43**, 7378–7411.
- 6 J. L. Zhang, J. Q. Zhong, J. D. Lin, W. P. Hu, K. Wu, G. Q. Xu, A. T. S. Wee and W. Chen, *Chem. Soc. Rev.*, 2015, **44**, 2998–3022.
- 7 G. Ke, C. Duan, F. Huang and X. Guo, *InfoMat*, 2020, **2**, 92–112.
- 8 V. Meded, A. Bagrets, K. Fink, R. Chandrasekar, M. Ruben, F. Evers, A. Bernand-Mantel, J. S. Seldenthuis, A. Beukman and H. S. J. van der Zant, *Phys. Rev. B*, 2011, **83**, 245415.
- 9 I. Diez-Perez, Z. H. Li, S. Y. Guo, C. Madden, H. L. Huang, Y. K. Che, X. M. Yang, L. Zang and N. J. Tao, *ACS Nano*, 2012, **6**, 7044–7052.
- 10 Y. Cao, S. H. Dong, S. Liu, Z. F. Liu and X. F. Guo, *Angew. Chem., Int. Ed.*, 2013, **52**, 3906–3910.
- 11 X. F. Guo, *Adv. Mater.*, 2013, **25**, 3397–3408.
- 12 M. L. Perrin, C. J. O. Verzijl, C. A. Martin, A. J. Shaikh, R. Eelkema, J. H. van Esch, J. M. van Ruitenbeek, J. M. Thijssen, H. S. J. van der Zant and D. Dulic, *Nat. Nanotechnol.*, 2013, **8**, 282–287.
- 13 V. Rabache, J. Chaste, P. Petit, M. L. Della Rocca, P. Martin, J. C. Lacroix, R. L. McCreery and P. Lafarge, *J. Am. Chem. Soc.*, 2013, **135**, 10218–10221.
- 14 B. Capozzi, Q. S. Chen, P. Darancet, M. Kotiuga, M. Buzzo, J. B. Neaton, C. Nuckolls and L. Venkataraman, *Nano Lett.*, 2014, **14**, 1400–1404.
- 15 P. Zhou, J. T. Zheng, T. Y. Han, L. J. Chen, W. Q. Cao, Y. X. Zhu, D. H. Zhou, R. H. Li, Y. Y. Tian, Z. T. Liu, J. Y. Liu and W. J. Hong, *Nanoscale*, 2021, **13**, 7600–7605.
- 16 D. Gust, T. A. Moore and A. L. Moore, *Chem. Commun.*, 2006, 1169–1178.
- 17 C. Schaefer, R. Eckel, R. Ros, J. Mattay and D. Anselmetti, *J. Am. Chem. Soc.*, 2007, **129**, 1488–1489.
- 18 C.-J. Xia, K. Gao, D.-H. Zhang, M. Yang and F.-L. Feng, *J. Appl. Phys.*, 2014, **116**, 083704.
- 19 T. Sendler, K. Luka-Guth, M. Wieser, J. Wolf, M. Helm, S. Gemming, J. Kerbusch, E. Scheer, T. Huhn and A. Erbe, *Adv. Sci.*, 2015, **2**, 1500017.
- 20 A. Vezzoli, R. J. Brooke, S. J. Higgins, W. Schwarzacher and R. J. Nichols, *Nano Lett.*, 2017, **17**, 6702–6707.
- 21 J. F. Zhou, K. Wang, B. Q. Xu and Y. Dubi, *J. Am. Chem. Soc.*, 2018, **140**, 70–73.
- 22 A. S. Kozlenko, I. V. Ozhogin, A. D. Pugachev, M. B. Lukyanova, I. M. El-Sewify and B. S. Lukyanov, *Top. Curr. Chem.*, 2023, **381**, 8.
- 23 L. Ma, G. Tian and J.-T. Lu, *Phys. Rev. B*, 2022, **106**, 165416.
- 24 P. Li, Y. Chen, B. Wang, M. Li, D. Xiang, C. Jia and X. Guo, *Opto-Electron. Adv.*, 2022, **5**, 210094.
- 25 Z. H. Li, M. Smeu, M. A. Ratner and E. Borguet, *J. Phys. Chem. C*, 2013, **117**, 14890–14898.

26 Z. H. Li, M. Smeu, S. Afsari, Y. J. Xing, M. A. Ratner and E. Borguet, *Angew. Chem., Int. Ed.*, 2014, **53**, 1098–1102.

27 M. Baghernejad, C. Van Dyck, J. Bergfield, D. R. Levine, A. Gubicza, J. D. Tovar, M. Calame, P. Broekmann and W. J. Hong, *Chem. – Eur. J.*, 2019, **25**, 15141–15146.

28 C. Zhan, G. Wang, X. G. Zhang, Z. H. Li, J. Y. Wei, Y. Si, Y. Yang, W. J. Hong and Z. Q. Tian, *Angew. Chem., Int. Ed.*, 2019, **58**, 14534–14538.

29 H. Audi, Y. Viero, N. Alwhaibi, Z. Chen, M. Iazykov, A. Heynderickx, F. Xiao, D. Guerin, C. Krzeminski, I. M. Grace, C. J. Lambert, O. Siri, D. Vuillaume, S. Lenfant and H. Klein, *Nanoscale*, 2020, **12**, 10127–10139.

30 C. L. Wu, A. Alqahtani, S. Sangtarash, A. Vezzoli, H. Sadeghi, C. M. Robertson, C. X. Cai, C. J. Lambert, S. J. Higgins and R. J. Nichols, *Nanoscale*, 2020, **12**, 7914–7920.

31 F. Yang, F. Chen, X. H. Wu, J. Luo, X. S. Zhou, J. R. Horsley, A. D. Abell, J. X. Yu, S. Jin and B. W. Mao, *J. Phys. Chem. C*, 2020, **124**, 8496–8503.

32 J. J. Parks, A. R. Champagne, T. A. Costi, W. W. Shum, A. N. Pasupathy, E. Neuscammman, S. Flores-Torres, P. S. Cornaglia, A. A. Aligia, C. A. Balseiro, G. K. L. Chan, H. D. Abruna and D. C. Ralph, *Science*, 2010, **328**, 1370–1373.

33 S. J. van der Molen and P. Liljeroth, *J. Phys.: Condens. Matter*, 2010, **22**, 133001.

34 F. Prins, M. Monrabal-Capilla, E. A. Osorio, E. Coronado and H. S. J. van der Zant, *Adv. Mater.*, 2011, **23**, 1545–1549.

35 S. Caneva, P. Gehring, V. M. Garcia-Suarez, A. Garcia-Fuente, D. Stefani, I. J. Olavarria-Contreras, J. Ferrer, C. Dekker and H. S. J. van der Zant, *Nat. Nanotechnol.*, 2018, **13**, 1126–1131.

36 M. C. Walkey, C. R. Peiris, S. Ciampi, A. C. Aragones, R. B. Dominguez-Espindola, D. Jago, T. Pulbrook, B. W. Skelton, A. N. Sobolev, I. D. Perez, M. J. Piggott, G. A. Koutsantonis and N. Darwish, *ACS Appl. Mater. Interfaces*, 2019, **11**, 36886–36894.

37 S. Kurihara, D. Yoneyama and T. Nonaka, *Chem. Mater.*, 2001, **13**, 2807–2812.

38 A. S. Kumar, T. Ye, T. Takami, B.-C. Yu, A. K. Flatt, J. M. Tour and P. S. Weiss, *Nano Lett.*, 2008, **8**, 1644–1648.

39 C. C. Huang, S. J. Chen, K. B. Ornsø, D. Reber, M. Baghernejad, Y. C. Fu, T. Wandlowski, S. Decurtins, W. J. Hong, K. S. Thygesen and S. X. Liu, *Angew. Chem., Int. Ed.*, 2015, **54**, 14304–14307.

40 M. J. Comstock, N. Levy, A. Kirakosian, J. Cho, F. Lauterwasser, J. H. Harvey, D. A. Strubbe, J. M. J. Fréchet, D. Trauner, S. G. Louie and M. F. Crommie, *Phys. Rev. Lett.*, 2007, **99**, 038301.

41 J. M. Mativetsky, G. Pace, M. Elbing, M. A. Rampi, M. Mayor and P. Samori, *J. Am. Chem. Soc.*, 2008, **130**, 9192–9193.

42 I. Kmentova, H. S. Sutherland, B. D. Palmer, A. Blaser, S. G. Franzblau, B. Wan, Y. Wang, Z. Ma, W. A. Denny and A. M. Thompson, *J. Med. Chem.*, 2010, **53**, 8421–8439.

43 M. M. Luebtow, L. Hahn, M. S. Haider and R. Luxenhofer, *J. Am. Chem. Soc.*, 2017, **139**, 10980–10983.

44 M. Sauer, K. H. Drexhage, U. Lieberwirth, R. Muller, S. Nord and C. Zander, *Chem. Phys. Lett.*, 1998, **284**, 153–163.

45 O. Sedlacek and R. Hoogenboom, *Adv. Ther.*, 2020, **3**, 1900168.

46 P. Beaujean, F. Bondu, A. Plaquet, J. Garcia-Amoros, J. Cusido, F. M. Raymo, F. Castet, V. Rodriguez and B. Champagne, *J. Am. Chem. Soc.*, 2016, **138**, 5052–5062.

47 M. Tomasulo and F. M. Raymo, *Org. Lett.*, 2005, **7**, 4633–4636.

48 M. Tomasulo, S. Sortino and F. M. Raymo, *Org. Lett.*, 2005, **7**, 1109–1112.

49 M. Tomasulo, S. Sortino, A. J. P. White and F. M. Raymo, *J. Org. Chem.*, 2005, **70**, 8180–8189.

50 J. Vogelsang, T. Cordes, C. Forthmann, C. Steinhauer and P. Tinnefeld, *Proc. Natl. Acad. Sci. U. S. A.*, 2009, **106**, 8107–8112.

51 L. Han, D. Iguchi, P. Gil, T. R. Heyl, V. M. Sedwick, C. R. Arza, S. Ohashi, D. J. Lacks and H. Ishida, *J. Phys. Chem. A*, 2017, **121**, 6269–6282.

52 M. Hintersteiner, A. Enz, P. Frey, A. L. Jaton, W. Kinzy, R. Kneuer, U. Neumann, M. Rudin, M. Staufenbiel, M. Stoeckli, K. H. Wiederhold and H. U. Gremlich, *Nat. Biotechnol.*, 2005, **23**, 577–583.

53 A. Laobuthee, S. Chirachanchai, H. Ishida and K. Tashiro, *J. Am. Chem. Soc.*, 2001, **123**, 9947–9955.

54 B. P. Mathew, A. Kumar, S. Sharma, P. K. Shukla and M. Nath, *Eur. J. Med. Chem.*, 2010, **45**, 1502–1507.

55 P. Panizzi, M. Nahrendorf, M. Wildgruber, P. Waterman, J.-L. Figueiredo, E. Aikawa, J. McCarthy, R. Weissleder and S. A. Hilderbrand, *J. Am. Chem. Soc.*, 2009, **131**, 15739–15744.

56 B. Zhao, K. Ranguelova, J. Jiang and R. P. Mason, *Free Radic. Biol. Med.*, 2011, **51**, 153–159.

57 Y. Zhang, P. Song, Q. Fu, M. Ruan and W. Xu, *Nat. Commun.*, 2014, **5**, 4238.

58 R. Usui, K. Yamamoto, H. Okajima, K. Mutoh, A. Sakamoto, J. Abe and Y. Kobayashi, *J. Am. Chem. Soc.*, 2020, **142**, 10132–10142.

59 Y. Wu, W. Liu, T. Liu, F. Li, T. Xiao, Q. Wu and Y. He, *ChemElectroChem*, 2018, **5**, 2171–2175.

60 M. Brandbyge, J. L. Mozos, P. Ordejon, J. Taylor and K. Stokbro, *Phys. Rev. B*, 2002, **65**, 165401.

61 Y. Q. Xue, S. Datta and M. A. Ratner, *Chem. Phys.*, 2002, **281**, 151–170.

62 J. Zhang, S. Hou, R. Li, Z. Qian, R. Han, Z. Shen, X. Zhao and Z. Xue, *Nanotechnology*, 2005, **16**, 3057–3063.

63 P. Hohenberg and W. Kohn, *Phys. Rev.*, 1964, **136**, B864–B871.

64 W. Kohn and L. J. Sham, *Phys. Rev.*, 1965, **140**, A1133–A1138.

65 Y. Cao, S. Dong, S. Liu, L. He, L. Gan, X. Yu, M. L. Steigerwald, X. Wu, Z. Liu and X. Guo, *Angew. Chem., Int. Ed.*, 2012, **51**, 12228–12232.

66 C. Jia, B. Ma, N. Xin and X. Guo, *Acc. Chem. Res.*, 2015, **48**, 2565–2575.

67 H. Sadeghi, S. Sangtarash and C. Lambert, *Nano Lett.*, 2017, **17**, 4611–4618.

68 X. F. Guo, J. P. Small, J. E. Klare, Y. L. Wang, M. S. Purewal, I. W. Tam, B. H. Hong, R. Caldwell, L. M. Huang, S. O'Brien, J. M. Yan, R. Breslow, S. J. Wind, J. Hone, P. Kim and C. Nuckolls, *Science*, 2006, **311**, 356–359.

69 Z. Qian, S. Hou, J. Ning, R. Li, Z. Shen, X. Zhao and Z. Xue, *J. Chem. Phys.*, 2007, **126**, 084705.

70 Y. Wang, H. Pan, D. Lin, S. Li, Y. Wang, S. Sanvito and S. Hou, *Phys. Chem. Chem. Phys.*, 2022, **24**, 21337–21347.

71 A. Garcia, N. Papior, A. Akhtar, E. Artacho, V. Blum, E. Bosoni, P. Brandimarte, M. Brandbyge, J. I. Cerdá, F. Corsetti, R. Cuadrado, V. Dikan, J. Ferrer, J. Gale, P. Garcia-Fernandez, V. M. Garcia-Suarez, S. Garcia, G. Huhs, S. Illera, R. Kortytar, P. Koval, I. Lebedeva, L. Lin, P. Lopez-Tarifa, S. G. Mayo, S. Mohr, P. Ordejon, A. Postnikov, Y. Pouillon, M. Pruneda, R. Robles, D. Sanchez-Portal, J. M. Soler, R. Ullah, V. W. Z. Yu and J. Junquera, *J. Chem. Phys.*, 2020, **152**, 204108.

72 J. M. Soler, E. Artacho, J. D. Gale, A. Garcia, J. Junquera, P. Ordejon and D. Sanchez-Portal, *J. Phys.: Condens. Matter*, 2002, **14**, 2745–2779.

73 N. Troullier and J. L. Martins, *Phys. Rev. B*, 1991, **43**, 1993–2006.

74 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.

75 A. R. Rocha, V. M. Garcia-Suarez, S. Bailey, C. Lambert, J. Ferrer and S. Sanvito, *Phys. Rev. B*, 2006, **73**, 085414.

76 J. Andzelm and E. Wimmer, *J. Chem. Phys.*, 1992, **96**, 1280–1303.

77 J. A. Pople, P. M. W. Gill and B. G. Johnson, *Chem. Phys. Lett.*, 1992, **199**, 557–560.

78 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, G. A. Petersson, H. Nakatsuji, X. Li, M. Caricato, A. V. Marenich, J. Bloino, B. G. Janesko, R. Gomperts, B. Mennucci, H. P. Hratchian, J. V. Ortiz, A. F. Izmaylov, J. L. Sonnenberg, D. Williams-Young, F. Ding, F. Lipparini, F. Egidi, J. Goings, B. Peng, A. Petrone, T. Henderson, D. Ranasinghe, V. G. Zakrzewski, J. Gao, N. Rega, G. Zheng, W. Liang, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, K. Throssell, J. A. Montgomery, Jr., J. E. Peralta, F. Ogliaro, M. J. Bearpark, J. J. Heyd, E. N. Brothers, K. N. Kudin, V. N. Staroverov, T. A. Keith, R. Kobayashi, J. Normand, K. Raghavachari, A. P. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, J. M. Millam, M. Klene, C. Adamo, R. Cammi, J. W. Ochterski, R. L. Martin, K. Morokuma, O. Farkas, J. B. Foresman and D. J. Fox, *Gaussian 16, Revision A.03*, Gaussian, Inc., Wallingford CT, 2016.

79 K. Fukui, *Acc. Chem. Res.*, 1981, **14**, 363–368.

80 C. Gonzalez and H. B. Schlegel, *J. Phys. Chem.*, 1990, **94**, 5523–5527.

81 R. Li, S. Hou, J. Zhang, Z. Qian, Z. Shen and X. Zhao, *J. Chem. Phys.*, 2006, **125**, 194113.

82 M. Paulsson and M. Brandbyge, *Phys. Rev. B*, 2007, **76**, 115117.

83 Z. Jiang, H. Wang, Y. Wang, S. Sanvito and S. Hou, *J. Phys. Chem. C*, 2017, **121**, 27344–27350.

84 Y. Li, X. Tu, H. Wang, S. Sanvito and S. Hou, *J. Chem. Phys.*, 2015, **142**, 164701.

85 S. Braun, W. R. Salaneck and M. Fahlman, *Adv. Mater.*, 2009, **21**, 1450–1472.

86 X. Crispin, V. Geskin, A. Crispin, J. Cornil, R. Lazzaroni, W. R. Salaneck and J. L. Bredas, *J. Am. Chem. Soc.*, 2002, **124**, 8131–8141.