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# Combining fracture mechanics and rheology to investigate the impact of micro-aeration on chocolate oral processing

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This study presents a rigorous mechanical characterisation investigation on milk chocolate with varying porosities, at different temperatures and strain rate levels. Uniaxial compression tests at temperatures varying from 20 °C to 30 °C were performed to measure the bulk properties of chocolate as a function of porosity and temperature. Fracture experiments were also conducted to compute the fracture energy at temperature levels between 20 °C and 30 °C for all tested samples. Additionally, rheological experiments are conducted to compute the viscosity of the different chocolates at 37 °C. This combined experimental analysis of solid mechanics, fracture mechanics, and rheology aims to define the impact of temperature and chocolate's phase change from solid to liquid on its mechanical properties. Moreover, the impact of micro-aeration on the relationship between material properties and temperature is discussed. The results demonstrate a significant impact of both temperature and micro-aeration on the chocolate's material properties; fracture stresses decrease with micro-aeration due to the presence of micro-pores creating weak links in the chocolate matrix, the critical strain energy release rate decreases with micro-aeration at temperatures up to 25 °C and increases at temperatures above 30 °C. Finally, the viscosity at 37 °C increases with increasing porosity due to the obstruction of the flow by micro-pores acting as "solid" particles. The results highlight how the impact of micro-aeration on the material properties of chocolate alters as the testing temperature rises and the material changes phase. The relationships between the micro-aeration and material properties and the dependence of temperature on the different mechanical properties are used to explain the difference in textural attributes as obtained from temporal dominance sensation tests. This study seeks to contribute valuable insights into the field of chocolate technology, emphasizing the need for a combined engineering approach to understand the structural breakdown of chocolate during oral processing as mechanisms such as chewing, melting, mixing and shearing occur.

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## 1. Introduction

A great effort in the food technology field has been demonstrated to shed insight into the structure–property relationship. The aim of this study is to optimise food processes such as industrial manufacturing and/or human oral process. Material properties as defined by engineering analysis are essential for controlling processes that involve mechanical loads. Flow processes are dictated by rheological properties<sup>1</sup> and processes which involve fracture, such as cutting and crushing, are

dominated by the fracture properties.<sup>2–5</sup> However, in processes such as food oral processing which involves simultaneous breakdown mechanisms, such as mastication, melting, and the shearing action of the tongue–palate interface, a combined engineering analysis is necessary to investigate the structure–property relationship. Such a rigorous study is notably missing from the literature.

In chocolate, rheological and fracture properties play a key role in the final quality of the product influencing the dispersion characteristics of chocolate's components during the manufacturing process while also affecting sensory characteristics, such as "hardness" and "smoothness" perception, and the characteristic snap.<sup>6–9</sup> Understanding and optimising these properties are crucial both for manufacturers, seeking to produce high-quality chocolate, and consumers, expecting a delightful sensory experience.

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There is a vast literature performing rheological experiments on chocolate materials attempting to establish the link between the manufacturing steps and rheological properties and also the structure–property relationship.<sup>10–12</sup> Servais *et al.*<sup>13</sup> proposed a methodology to measure the rheology of chocolate and related cocoa products using factory grade rheometers. Afoakwa *et al.*<sup>14</sup> provided a detailed review of factors influencing the rheological properties of chocolate, concluding that the processing technique, the ingredients and the particle size distribution are the most important factors. Vázquez *et al.*<sup>15</sup> performed a rheological study on four commercial chocolate samples with similar microstructures and reported a shear thinning behaviour. According to Windhab,<sup>16</sup> melted chocolate is a non-Newtonian fluid as the viscosity depends on the temperature, shear rate, and time. The viscosity measurements are linked to sensorial attributes such as the “grittiness” perception, and based on the same study, chocolate is considered as “smooth” for viscosity values within 1.5–3.5 Pa s. These values are recorded at a shear rate of 20 s<sup>−1</sup> and at a temperature of 37 °C, *i.e.*, human body temperature. Melted chocolate contains sugar, cocoa and milk particles dissolved in the melted fat matrix. Lannes *et al.*<sup>17</sup> studied the flow behaviour of chocolate drinks at 25 °C and 40 °C and found that Bingham plastic models fit well with the experimental data. The rheological behaviour of coconut flavoured milk with varying fat content (15–30%) was investigated in the study by Simuang *et al.*<sup>18</sup> at temperatures ranging between 70 °C and 90 °C. They reported pseudoplastic behaviour, and the effect of temperature on the apparent viscosity was captured using the Arrhenius equation.

While considerable effort is focused on measuring the rheological properties of chocolate, the literature narrows significantly when the focus shifts to the critical strain energy release rate,  $G_c$ , of chocolate. Based on the authors' knowledge, in a few studies where  $G_c$  of chocolate was investigated, a large deviation in the reported values is observed. Namely, Chen *et al.* utilised wedge-cutting tests on chocolate with different compositions, reporting a wide range of  $G_c$  values between 330 and 1189 J m<sup>−2</sup>.<sup>19</sup> Gomez *et al.* proposed a numerical method to inversely calculate the critical strain energy release rate of chocolate through SENB simulations and reported a value of 2.1 J m<sup>−2</sup>.<sup>20</sup> Even though translating the fracture energy of chocolate from Charpy tests into the critical strain energy release rate might be debatable, Parsons and Goodall reported absorbed energy on fracture which varied from 70% to 78% depending on the type of chocolate. In our previous work on the mechanical properties of chocolate, values of  $G_c$  for non-aerated milk chocolate at 20 °C ranged from 8 to 22 J m<sup>−2</sup>, showing a dependency on the testing speed.<sup>21,22</sup>

The importance of measuring the critical strain energy release rate of chocolate extends beyond being a key input computational parameter for simulating various processes with industrial significance;<sup>2,23,24</sup> it is also intricately linked to consumer perception.<sup>4,25,26</sup> Based on previous studies,<sup>9,20,21,27</sup> chocolate hardness, affected both by the strength and toughness of the food, is linked to the early stages of oral processing.

It guides the subsequent steps of fragmentation, chocolate melting, and the shearing action between the oral surfaces.<sup>27–31</sup> However, as the process progresses and a combination of sensations is perceived, rheological properties begin to dominate the process.<sup>30,32</sup> The critical point where the shift of dominance is observed is unknown.

Therefore, we believe that the fracture and rheological properties of chocolate should be studied together and as a function of temperature. This approach is essential for understanding the mechanisms occurring during oral processing and their effects on the sensorial attributes perceived at these intermediate steps of oral processing. To unravel the underlying mechanisms in a chocolate system, a rigorous experimental campaign must be undertaken.

To the best of the authors' knowledge, no such study in the literature has been reported to date. Integrating these two experimental methodologies aims to provide a holistic understanding of the mechanical and rheological aspects governing chocolate quality.

In our previously published work on the effect of micro-aeration, the authors presented a comprehensive experimental analysis to calculate the mechanical properties of chocolate at 20 °C under different loading scenarios and proposed a novel multiscale computational tool to simulate the first bite of chocolate.<sup>3,22,33,34</sup> However, the effect of temperature on the mechanical properties of chocolate at temperatures relevant to oral processing conditions, 20–37 °C, which is important after the first bite, remained to be investigated. Additionally, the impact of micro-aeration on the rheological properties of chocolate is lacking in the literature too. While an initial attempt to link the mechanical properties of chocolate to explain the mechanisms happening during oral processing was performed,<sup>27</sup> the lack of the dependency of chocolate's mechanical response on the temperature and the coupling to rheological analysis prevent a more comprehensive explanation of the oral processing mechanisms as observed from temporal dominance sensation (TDS) tests.

Therefore, this study delves into the combined exploration of fracture and rheological properties in chocolate and attempts to use this combined analysis to explain the temporal dominance mechanisms occurring during food oral processing of chocolate. More specifically, single-edge notched bending (SENB) tests at 20 °C and 25 °C are performed to study the fracture behaviour of chocolate when it is considered to be in its solid regime. Wire-cutting tests are conducted at 30 °C when the chocolate is in a soft state. On the other hand, when chocolate is considered fully melted, rotational experiments are conducted at shear rates between 0.1 and 1000 s<sup>−1</sup>. Finally, combined engineering analysis is performed to discuss temporal dominance sensation test data and explain the evolving mechanisms occurring during oral processing.

## 2. Materials & methods

### 2.1. Materials

Three milk chocolate materials with typical compositions and different levels of micro-aeration, 0%, 10%, and 15%, were



provided by the Nestlé Product Technology Centre (NPTC) in York, UK. The composition of all samples was as follows: 44 wt% of sugar crystals, 27 wt% of cocoa fats, 10 wt% of milk powders, 6 wt% of cocoa solids and a small percentage of sunflower lecithin, *i.e.*  $\sim 0.3$  wt%. Details regarding the manufacturing steps of the chocolate samples and the micro-aeration process are reported in our previous published work.<sup>21,22</sup> The chocolate samples used in the uniaxial compression and fracture experiments were provided in the following geometries: cylinders of 20 mm diameter,  $D_0$ , and 20 mm height,  $L_0$ , and rectangular beams of 80 mm length,  $L$ , 15 mm width,  $W$ , and 8 mm thickness,  $B$ , as shown in Fig. 2a and c, respectively.

## 2.2. Chocolate microstructure

The scanning electron microscopy (SEM) micrographs of all chocolate materials with different levels of micro-aeration are shown in Fig. 1, as obtained by Bikos *et al.*<sup>21,22</sup> Based on these figures, the differentiating factor between the chocolate microstructures is the presence of micropores, which, according to our previous published work, have an average diameter of approximately 40–45  $\mu\text{m}$  and can be up to approximately 100  $\mu\text{m}$ . This size is fairly similar (10–40  $\mu\text{m}$ ) to other microscopic features found in chocolate, such as sugar crystals and cocoa solids.<sup>35</sup> For further details about the size distribution of pores and the microscope testing parameters, the reader is referred to Bikos *et al.*<sup>21,22</sup>

## 2.3. Experimental methods

**2.3.1. Unconfined uniaxial compression tests.** Unconfined monotonic uniaxial compression experiments were performed for all chocolate materials on a 5584 universal Instron testing instrument with a 2 kN load cell at varying temperature and strain rate conditions. More specifically, three temperatures were selected, namely 20 °C, 25 °C, and 30 °C, and three applied true strain rates (the true specimen's height was recorded for every repeat), *i.e.*  $\dot{\epsilon} = 0.01 \text{ s}^{-1}$ ,  $0.1 \text{ s}^{-1}$  and  $1 \text{ s}^{-1}$ . These tests were performed under monotonically increasing applied strain up to fracture. The three temperature thresholds were selected to highlight the change in the chocolate's mechanical behaviour at room temperature, at a temperature just before the phase

change, and temperature where the phase change occurred<sup>18,36</sup> and correlate the response with rheological experiments under shear when the chocolate is fully melted at 37 °C. The tests were not extended beyond 30 °C because of the response of the chocolate in a liquid state which violates the applicability of the mechanical testing protocols.

To control the temperature of the chocolate during the compression tests, a temperature chamber was fitted to the 5584 Instron machine. The chamber (Fig. 2) was able to monitor the temperature within the chamber with an accuracy of 0.1 °C. Due to the relatively low thermal conductivity of the chocolate,<sup>37,38</sup> approximately 60 minutes were allowed for the temperature to equilibrate, based on thermocouple indications on a “dummy” chocolate specimen placed inside the chamber.

A thin PTFE film was applied on the compression plates to minimise the friction coefficient at the chocolate–compression plate interface.<sup>39</sup> A Logitech HD Pro Webcam C920 was used to monitor the chocolate samples during compression tests and ensure that no “barrelling” mechanism was apparent, which could interfere with the accurate computation of the chocolate's material properties. In addition, preliminary tests on chocolate samples (results not shown) with different aspect ratios showed no significant variations providing further evidence for the accuracy of the calculated material properties. Compliance tests were conducted at the same strain rates by bringing the compression plates in contact with each other to measure the deflection which is not associated with the material deformation. Preliminary results showed no significant machine compliance effect in the computed  $\sigma$ – $\epsilon$  response, *i.e.* deviations less than 5%.

The true stress,  $\sigma$ , and strain,  $\epsilon$ , for a uniaxial compression test are calculated as follows:

$$\sigma = \frac{F}{A_i} \quad (1)$$

$$\epsilon = \ln \frac{H_i}{H_o} \quad (2)$$

where  $F$  is the applied force as obtained from the testing instrument,  $H_i$  and  $H_o$  are the current and original (undeformed) sample height, respectively, and  $A_i$  and  $A_o$  are the current and the original (undeformed) cross-sectional areas of

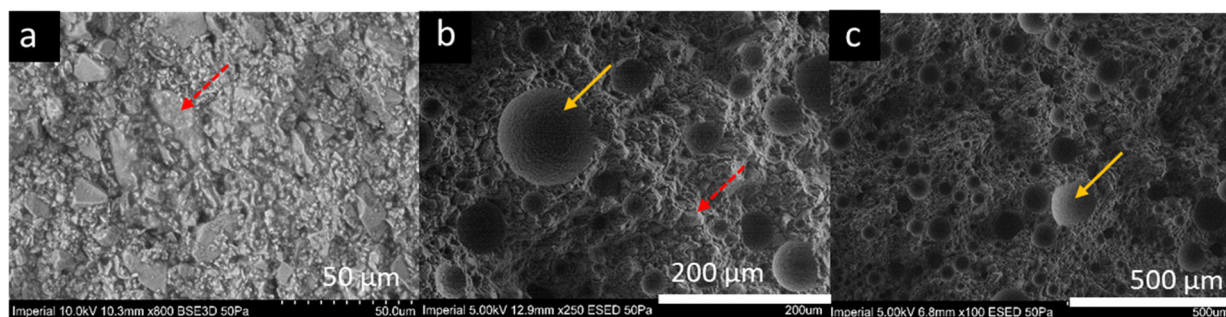


Fig. 1 Scanning electron microscopy (SEM) micrographs of the (a) non-aerated, (b) 10%, and (c) 15% micro-aerated chocolate under different magnifications. The red and yellow arrows depict sugar particles and micro-pores, respectively. The images of the micrographs are modified from Bikos *et al.*<sup>22</sup>





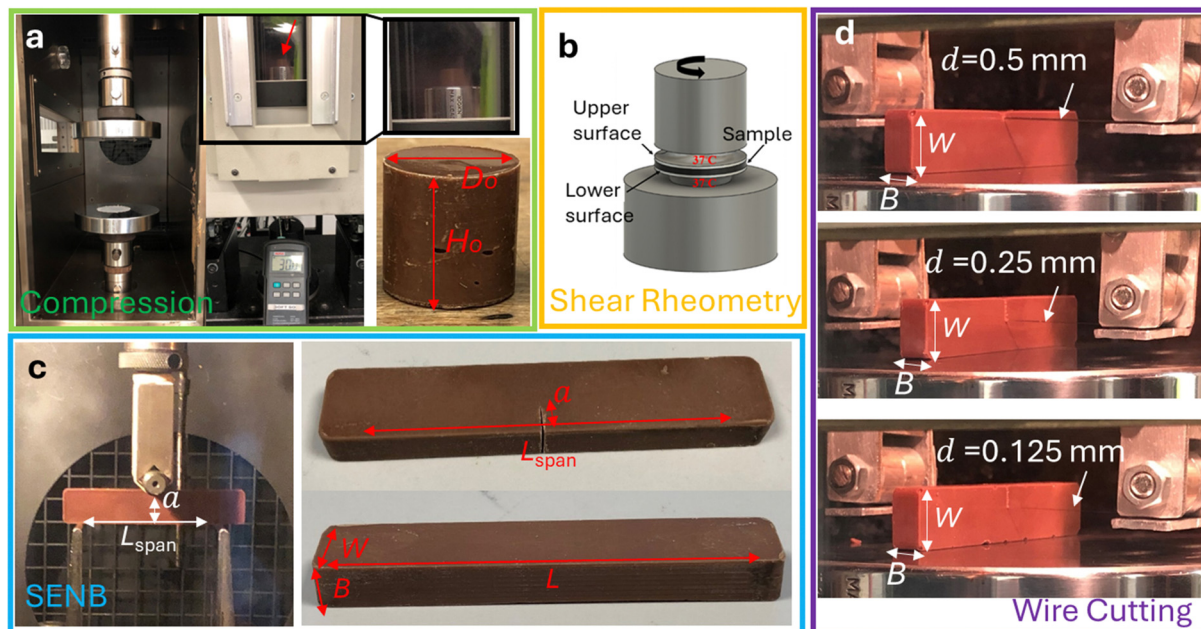


Fig. 2 Overview of the experimental setup and specimen geometry from the different experimental methods. Image (a) shows the temperature chamber installed on the Instron machine together with the geometry of the compression specimen, (b) the rheological setup, (c) the SENB setup with the notched beam specimen installed and geometry of the notched and unnotched beam specimen, and (d) the wire cutting setup with the different wire diameters installed on the holder.

the sample. The fact that chocolate indicated a Poisson's ratio close to 0.49<sup>37</sup> confirms the assumption of incompressible behaviour. Based on this assumption, the current sample cross-sectional area,  $A_i$ , is calculated using the equation below:

$$H_i A_i = H_o A_o \quad (3)$$

The compressive modulus under monotonic conditions is measured from the initial linear part of the true  $\sigma$ - $\epsilon$  curve. Based on previously published work on chocolate,<sup>22,33</sup> the modulus computed from the initial linear region does not correspond to the true Young's modulus, which should be computed from the initial linear part from the unloading region of the  $\sigma$ - $\epsilon$  curve from a loading unloading experiment. In this work, the monotonic compression tests are employed to establish a connection between the mechanical response of the chocolate in a solid state, through compression tests, and a liquid state, through a shear rheometer. In addition, since the first bite regards a monotonic loading process, the initial apparent modulus from monotonic compression tests might be more relevant. For this reason, the modulus measured in this study will be referred to as the apparent Young's modulus. The yield stress was computed as a cut-off point ("plastic knee") when the stress started to level off and is referred to throughout the text as macroscopic yield stress. The compressive strength was measured as the maximum point in the stress-strain curve, which based on preliminary results coincided with the fracture point. Therefore, in this text, the fracture stress corresponds to the compressive strength as well.

Uniaxial compression tests were preferred over tensile and bending experiments because they are closely associated with

the compressive actions of the teeth against the food during mastication. Previous reports have shown the complex mechanical behaviour of the chocolate, which is dependent on the mode of deformation, *i.e.*, tension, compression, and bending.<sup>33</sup> The results of this study highlighted complex microscopic mechanisms that cause differences in the computed stress-strain curve and the bulk mechanical properties of the chocolate depending on the loading scenario. Although loading-unloading experiments are recommended in previous studies to compute an accurate value of Young's modulus, they are not directly relevant here. This study aims to use engineering analysis to explain changes in consumer perception due to micro-aeration during oral processing. The first bite and mastication are a monotonic process, and hence, employing uniaxial unconfined compression is more appropriate.

**2.3.2. Rheological experiments.** The rheological experiments were carried out using a Discovery Hybrid Rheometer, model HR-1 (TA Instruments, USA), a schematic representation of which is shown in Fig. 2b. Cross hatch parallel plates of 50 mm radius were used with a gap of 1000  $\mu\text{m}$  in order to prevent the wall slip effects in steady state flow sweep measurements. The chocolate samples were pre-melted in an oven at 40  $^{\circ}\text{C}$  and then transferred to the plates. The rheometer plates were heated to 37  $^{\circ}\text{C}$ , to mimic the physiological conditions in the mouth<sup>28</sup> and kept constant for the duration of the tests. Initially, it was attempted to melt the chocolate specimens directly on the rheometer plates at 37  $^{\circ}\text{C}$ , but separation during the rheological test was observed implying that the chocolate was not fully melted. Flow sweep tests were performed at an increasing shear rate from 0.01–1000  $\text{s}^{-1}$ . The flow curve



was then obtained by plotting the viscosity as a function of the applied shear rate. Before each test, the upper and lower plates were removed, cleaned and carefully placed onto the rheometer. Afterwards, the upper and lower plates were lowered slowly until they came into contact with each other, and a small normal force ( $\sim 0.1$  N) was recorded. The apparatus was then calibrated by setting the gap between the parallel plates to zero and the test was ready to start. The effect of saliva on the rheological properties of chocolate was also investigated. The flat plates were submerged in artificial saliva for 60 minutes allowing the proteins to be absorbed. By submerging the flat plates in artificial saliva, it is assumed that due to the short time scale of the final stage of the oral process, the artificial saliva will form a layer on the outer surface of the chocolate. The same artificial saliva recipe was used in our previous published work where the tribological properties of micro-aerated chocolates were studied.<sup>28</sup> The artificial saliva-infused parts were then heated to 37 °C and the tests were started.

The Carreau<sup>40</sup> and Cross<sup>41</sup> models will be used to characterise the flow behaviour of the chocolate samples:

$$\eta = \eta_{\infty} + \frac{\eta_0 - \eta_{\infty}}{[1 + (\lambda_c \dot{\gamma})^2]^p} \quad (4)$$

$$\eta = \eta_{\infty} + \frac{\eta_0 - \eta_{\infty}}{1 + (\alpha_c \dot{\gamma})^m} \quad (5)$$

where  $\eta_0$  is the constant zero-shear viscosity,  $\eta_{\infty}$  is the infinite-shear-viscosity value,  $\dot{\gamma}$  is the shear rate,  $\lambda_c$  and  $\alpha_c$  are time constants, and  $p$  and  $m$  are dimensionless exponents for the Carreau and Cross models, respectively.<sup>15,40</sup> It is worth noting that the widely used Casson model was also examined and calibrated with the experimental data, but it was not found applicable for the specific chocolate materials.

**2.3.3. Fracture experiments.** SENB tests, conducted according to the ASTM standard D5045-99,<sup>42</sup> were performed on all chocolate materials to measure its critical strain energy release rate,  $G_c$ , using a three-point bending setup on a 5584 Instron testing instrument with a load cell capacity of 100 N (maximum load observed was approximately 20 N) under the following environmental conditions: 20 °C temperature and 50% humidity. To ensure plane strain conditions, the length span (referred to as “ $L_{\text{span}}$ ” in Fig. 2c) and the notch length across the width (referred to as “ $a$ ” in Fig. 2c) were chosen to be 60 mm and 7 mm, respectively, aligning with the ASTM protocol requirements and  $a/W$  ( $0.45 \leq a/W \leq 0.55$ ). Each chocolate sample was notched using a Zona SA35/05 ultra-thin razor saw, and the developed notch was pre-cracked using a new steel razor blade (RS Components Ltd) for each test. A tap with a plastic mallet was applied to ensure a sharp natural crack at the tip of the notch.

Force-displacement data were recorded at three constant crosshead speeds: 0.02 mm s<sup>-1</sup>, 0.2 mm s<sup>-1</sup> and 2 mm s<sup>-1</sup>. In addition, force-displacement data obtained from the machine compliance experiments were used to eliminate any indentation or compliance effects from the instrument. To consider these effects, unnotched chocolate specimens

(see Fig. 2c) were placed on a three-point bending setup. The support pins were moved together to the mid-span location under the loading pin, and the chocolate was loaded up to the same load level as the SENB experiments. Then, for a given load, the displacement recorded from the machine compliance experiment was subtracted from the one measured from the SENB experiments.<sup>42</sup> Four repeats were conducted for each testing speed for all chocolate materials. The same experimental procedure was repeated at 25 °C for all types of chocolate and all applied strain rates. To ensure uniform temperature throughout the samples and control the temperature conditions, the same temperature chamber as the one used in the compression experiments was installed in the testing machine, as shown in Fig. 2c, and the samples were left for an hour for temperature equilibration. Similarly, four repeats were used for each testing condition and all types of chocolate.

Due to the soft nature of the chocolate at elevated temperatures, wire cutting experiments were performed on all materials (0%, 10%, and 15% micro-aeration) at 30 °C and 50% relative humidity to measure the corresponding critical strain energy release rate,  $G_c$ . Similar to the other mechanical tests, a temperature chamber was placed in the testing instrument to control the set temperature. The wire was attached to a holder and was adjusted to be tight enough to prevent any significant deflections. The holder with the wire attached to it was placed on a 100 N load cell. The wire cutting test configuration is shown in Fig. 2d.

The cutting force-displacement data were recorded at three constant crosshead speeds of 0.02 mm s<sup>-1</sup>, 0.2 mm s<sup>-1</sup> and 2 mm s<sup>-1</sup> using three different wire diameters of 0.5 mm, 0.25 mm, and 0.125 mm for each speed. Preliminary wire cutting experiments ensured that the wire remained straight and tight throughout the cutting process, with no excess material concentrated at the crack tip that could interfere with the force-displacement measurements. Three repeats were performed for each experimental condition, *i.e.* testing speed, wire diameter, and type of material. This study did not investigate the effect of the sample dimensions, such as width and length, since the current dimensions ( $L, B > 10d$ ) meet the requirements for reproducible wire cutting forces.<sup>2,43</sup> The  $G_c$  data and their error bars were obtained using the statistical methodology presented by Williams and Rink.<sup>44</sup>

The critical strain energy release rate is evaluated by recording the area underneath the force-displacement curve and assuming linear elastic behaviour. Based on Linear Elastic Fracture Mechanics theory, the critical energy release rate  $G_c$  is computed from

$$G_c = \frac{U}{BW\phi} \quad (6)$$

where  $\phi$  is an energy calibration factor depending on the ratio  $a/W$ , which is given in the ASTM standards,<sup>42</sup> and  $U$  is the energy calculated from the area under the force-displacement curve.



The computation of  $G_c$ , performed using eqn (6), was performed as a function of the crosshead speed to highlight potential rate-dependency. The values of  $G_c$  in these tests refer to the measurements obtained during crack initiation, *i.e.*, initiation  $G_c$ . Due to the challenges associated with measuring accurately crack growth during SENB tests, the propagation critical strain energy release rate was not calculated in this study.

For a direct comparison with the  $G_c$  data obtained from the SENB experiments, the  $G_c$  measurements from wire cutting tests were plotted as a function of the crack speed,  $\dot{a}$ . For wire cutting experiments, the crack speed equals the wire cutting speed. However, the crosshead speed,  $\dot{u}$ , during the SENB tests is not equal to the crack speed,  $\dot{a}$ , in front of the crack tip. To convert the crosshead speed,  $\dot{u}$ , into the crack speed,  $\dot{a}$ , during SENB experiments, the methodology introduced by Gamonpi-las *et al.* for starch gels is employed.<sup>45</sup>

Assuming that  $G_c$  is constant during crack propagation and the crack growth is stable, which is true for large crack lengths ( $\sim a/W > 0.4$ ), the crack speed in front of the crack,  $\dot{a}$ , is measured through

$$\dot{a} = \frac{X}{t} W \quad (7)$$

where  $t$  is the time corresponding to the crack initiation and  $X$  is evaluated from<sup>45</sup>

$$X = \frac{2\phi_1 x}{x^{1+m_1} + m_1 \phi_1} \quad (8)$$

where  $\phi_1$  and  $m_1$  are constants that depend on the ratio of  $L_{\text{span}}/W$ , and they are equal to 0.142 and 0.787, respectively, for  $L_{\text{span}}/W = 4$ . For relatively larger crack lengths, where the crack growth is stable from the beginning of the crack growth,  $x$  corresponds to the initial  $x_0 = a/W$ . In this study,  $x_0$  was controlled to be within the suggested range from the ASTM standards:<sup>42</sup>  $0.45 \leq a/W \leq 0.55$ .

The crack speed values measured from eqn (7) are utilised to highlight the dependency of  $G_c$  from the SENB tests on the crack speed, enabling a comparison to  $G_c$  data derived from the wire cutting experiments.

The methodology to calculate  $G_c$  from wire cutting experiments is based on the assumption that multiple dissipation mechanisms occur in front of the crack tip. These mechanisms include the energy associated with the plastic deformation during cutting, friction energy present during cutting, and energy associated with producing a new surface. The resultant stresses associated with these mechanisms are illustrated in Fig. 3a.

In a wire cutting process, the applied energy increases with the increasing wire diameter. However, this increase in applied energy is primarily caused by the rise in plastic deformation and friction energy due to the higher volume of material in contact with the wire. By using wires of varying diameters, extrapolating the steady state cutting force per unit thickness ( $B$ ), corresponding to different wire diameters  $d$ , to a zero diameter provides an estimation of the fracture energy,  $G_c$ . Typical data extracted from such experiments are summarised in Fig. 3b.

The methodology assumes a material with an elastic-plastic stress-strain response with no strain softening or hardening. However, the methodology has also been successfully applied to other materials with a strain hardening response similar to that shown for the chocolate in this study.<sup>46</sup> Assuming yielding occurs at the crack tip, the cutting force ( $F$ ) is associated with the yield stress ( $\sigma_y$ ), coefficient of friction ( $\mu_f$ ), and wire diameter ( $d$ ), as per:<sup>46</sup>

$$F dx = \int_0^{\pi/2} B d (\sigma_y \cos \theta + \mu_f \sigma_y \sin \theta) d\theta \quad (9)$$

In addition, the energy release rate required for fracture, the critical strain energy release rate,  $G_c$ , is given by

$$F dx = B G_c dx \quad (10)$$

Adding eqn (9) and (10), a linear expression between  $F/B$  and  $d$  is derived. The y-axis intercept of a plot  $F/B-d$  would then be equal to  $G_c$ .

For a given yield stress,  $\sigma_y$ , the friction coefficient,  $\mu_f$ , can also be estimated through the slope of this linear fit to the experimental data.

$$\frac{F}{B} = G_c + (1 + \mu_f) \sigma_y d \quad (11)$$

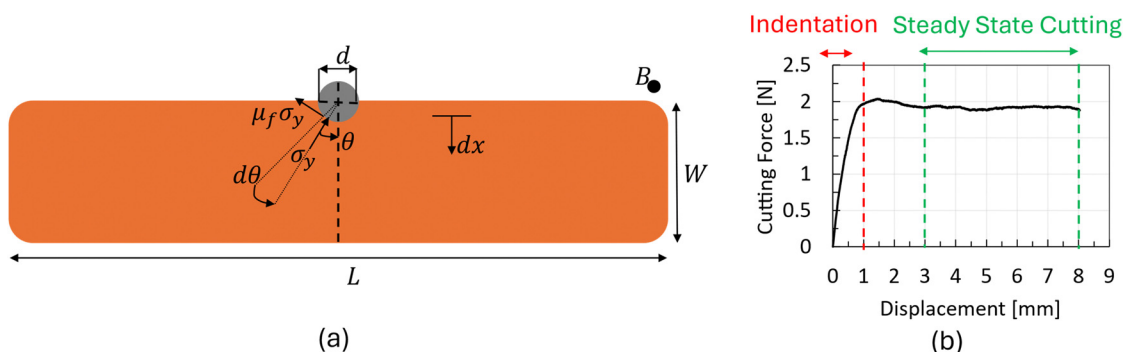


Fig. 3 (a) Schematic of the beam specimen showing the cutting parameters from eqn (11) and (b) a typical load-displacement curve extracted from the wire cutting test indicating the indentation and the steady state cutting state.



Note that eqn (11) refers to rate-independent materials. To account for rate effects, the terms  $G_c$  and  $\sigma_y$  are replaced as follows:<sup>2</sup>

$$G_c = G_o \frac{\dot{\epsilon}^{n_w}}{\dot{\epsilon}_o} \quad (12)$$

$$\sigma_y = \sigma_o \frac{\dot{\epsilon}^{m_w}}{\dot{\epsilon}_o} \quad (13)$$

where  $\dot{\epsilon}$  is the strain rate and  $\dot{\epsilon}_o$  is a parameter taken to be  $1 \text{ s}^{-1}$  and is used to make the ratio  $\frac{\dot{\epsilon}}{\dot{\epsilon}_o}$  dimensionless. The parameters  $n_w$  and  $m_w$  are power law constants. In addition,  $G_o$  refers to the critical strain energy release rate at the strain rate  $\dot{\epsilon}_o$  and  $\sigma_o$  is the stress responsible for the energies associated with the viscous dissipation and friction at the same strain rate,  $\dot{\epsilon}_o$ .<sup>43</sup> Note that the power law constant  $m_w$  is obtained by creating isometric curves, *i.e.*, stress,  $\sigma$ , *vs.* strain rate,  $\dot{\epsilon}$ , from compression experiments for different strain intervals. Similarly, the constant  $n_w$  is obtained by extracting the slope from cutting energies,  $F/B$ , *vs.* crack speed,  $\dot{a}$ , plots for different wire diameters.<sup>2</sup> Based on preliminary results on the tested materials,  $n_w$  and  $m_w$  were found to be approximately 0.04, 0.06, and 0.05 and 0.13, 0.14, and 0.13 for the 0%, 10%, and 15% micro-aerated chocolate, respectively. Goh *et al.*, who investigated the fracture behaviour of different types of cheese, reported that  $n_w = m_w$ . The values of these constants ranged between 0.16 and 0.19 for different types of cheese materials.<sup>2</sup>

The strain rate,  $\dot{\epsilon}$ , is related to the wire diameter,  $d$ , and crack speed  $\dot{a}$ , through

$$\dot{\epsilon} = \left( \frac{K\dot{a}}{(d+C)} \right) \quad (14)$$

where  $C$  and  $K$  are constants, which according to Goh *et al.*'s study on cheese, they are only geometric parameters.<sup>2</sup> Therefore, the same values for  $C$  and  $K$  as the study on cheese were employed, namely 0.14 mm and 0.48, respectively.<sup>2</sup>

Substituting eqn (12)–(14), into eqn (11) yields:

$$\frac{F}{B} \left( \frac{\dot{\epsilon}_o(d+C)}{K\dot{a}} \right)^{m_w} = G_o + (1 + \mu_f) \sigma_o \left( \frac{\dot{\epsilon}_o(d+C)}{K\dot{a}} \right)^{m_w - n_w} d \quad (15)$$

The critical strain energy release rate for different speeds is computed by the intercept of the plot  $F/B \left( \frac{\dot{\epsilon}_o(d+C)}{K\dot{a}} \right)^{m_w}$  versus  $\left( \frac{\dot{\epsilon}_o(d+C)}{K\dot{a}} \right)^{m_w - n_w} d$ .

### 3. Experimental results

#### 3.1. Uniaxial compression results

The stress–strain curves for all chocolate materials at temperatures of 20 °C, 25 °C, and 30 °C, under three strain rates, are shown in Fig. 4. Each curve in this figure exhibits a small variability amongst the repeats, with a maximum error of less than 10% (the error bars at the highest strain rate are

included). The red cross-shaped marks in the plots signify the strain at ultimate failure. Except for the curves at 30 °C, the strain rate dependency is observed in the nonlinear region, *i.e.*, initiating from the onset of macroscopic yielding. However, at 30 °C, significant rate dependency is observed from the beginning of the material's deformation. As shown in Fig. 4a, a strain hardening effect is present in the stress–strain response of the non-aerated chocolate, *i.e.*, the stresses increase after yielding initiation with increasing applied strain (see the red curve in Fig. 4a). This effect diminishes with increasing porosity and increasing strain rate, see Fig. 4a–c.

Assessing the effect of chocolate micro-aeration on its stress–strain response, an overall reduction of stresses with increasing porosity is observed. This is true regardless of the testing conditions such as temperature and strain rate. This observation is consistent with results in the literature, reporting that pores create weaker material structures compared to their solid counterparts.<sup>47–49</sup>

Similarly, the effect of temperature on the stress–strain curves leads to a reduction in the stresses too, see Fig. 4d, e, and f. Especially at 30 °C (Fig. 4g–i), the stresses are an order of magnitude lower compared to the values at lower temperatures, providing further evidence of the transition of chocolate to a soft solid at these conditions. Interestingly, at 30 °C, and in contrast to the stress–strain behaviour at 20 °C and 25 °C, a non-linear, rate-dependent response is noted without a distinct linear region. In such cases, the computation of the Young's modulus becomes a subjective exercise but not essential too.

For a more quantitative comparison, the macroscopic yield stress,  $\sigma_y$ , and the fracture stress,  $\sigma_f$ , are plotted as a function of strain rate for chocolate materials at 20 °C as shown in Fig. 5a and b. According to these figures, both material properties decrease with increasing porosity. For all testing speeds, an average difference of 18% is observed between the yield stress values of the non-aerated chocolate and the 10% micro-aerated chocolate. This difference increases to 27% when comparing the macroscopic yield stress of the non-aerated chocolate to the macroscopic yield stress of the 15% micro-aerated chocolate. Similar decreasing trends are observed for the fracture stress values. These decreasing trends of stresses with increasing porosity are common in porous materials such as porous metals, polymers, and cement.<sup>47–49</sup>

Fig. 5c and d summarise the data for the macroscopic yield stress and the fracture stress for the three chocolate materials as a function of strain rate at 25 °C. Both the macroscopic yield stress and the fracture stress exhibit significant rate-dependent response and they also decrease with increasing porosity. A maximum difference of 19% (at  $\dot{\epsilon} = 0.1 \text{ s}^{-1}$ ) in macroscopic yield stress is observed when comparing the non-aerated chocolate to the 10% micro-aerated chocolate. The maximum difference increases to 42% when the macroscopic yield stress of the non-aerated chocolate is compared to the macroscopic yield stress of the 15% micro-aerated chocolate, with the maximum difference occurring at a strain rate of  $0.1 \text{ s}^{-1}$ . Similar comparisons for the fracture stress values result in differences of 22% ( $\dot{\epsilon} = 0.1 \text{ s}^{-1}$ ) and 46% ( $\dot{\epsilon} = 0.1 \text{ s}^{-1}$ ), respectively.





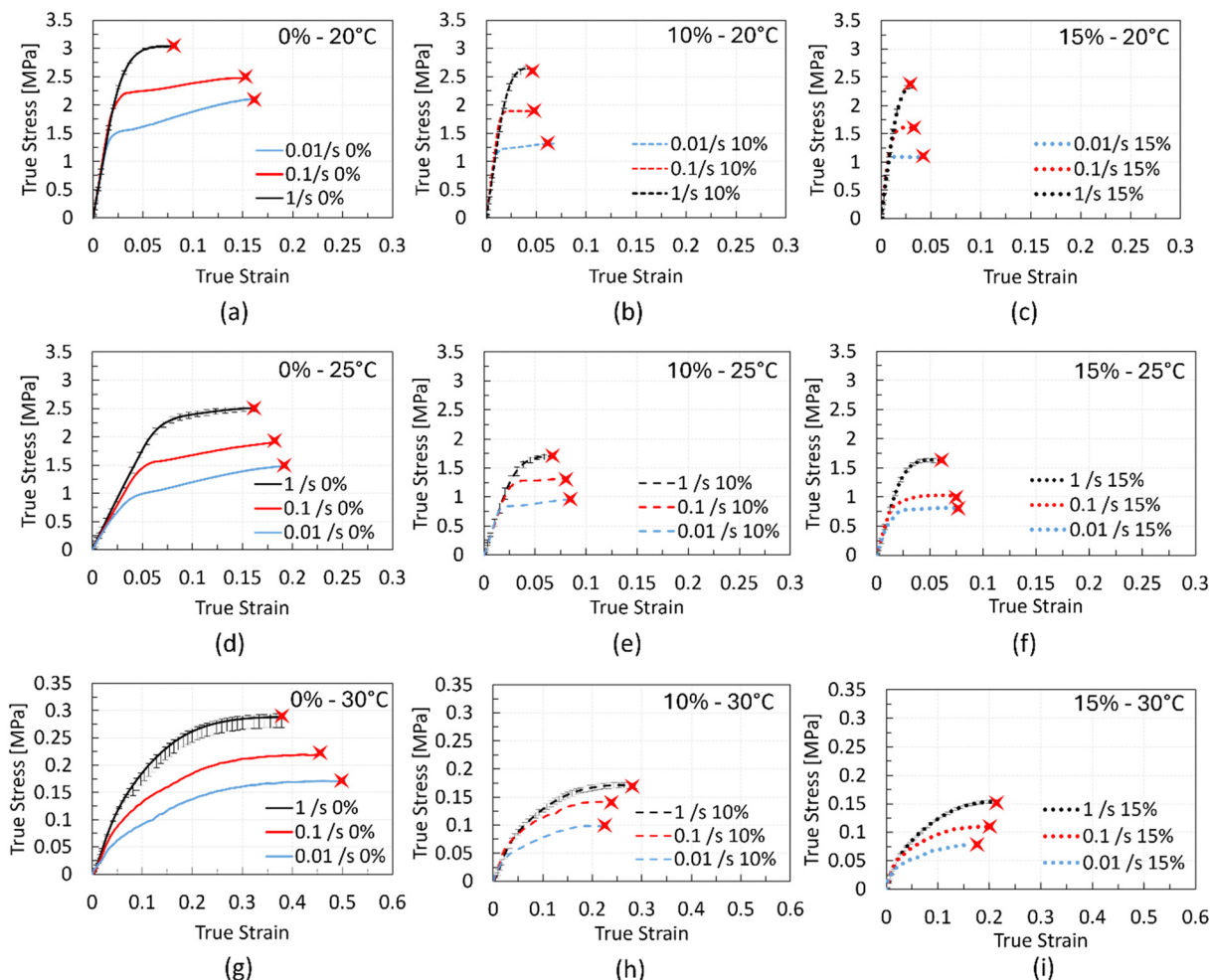


Fig. 4  $\sigma$ - $\epsilon$  data for chocolate samples of different porosities from the uniaxial compression tests under three strain rates at 20 °C, 25 °C, and 30 °C. The red "x" marks indicate the location where ultimate failure occurs.

Assessing the impact of micro-aeration on the macroscopic yield stress,  $\sigma_y$ , and fracture stress,  $\sigma_f$ , when the chocolate reaches a temperature of 30 °C, an interesting phenomenon is observed. Based on Fig. 5f, the values of  $\sigma_f$  at 30 °C indicate no significant difference between the micro-aerated products, *i.e.*, the error bars of these measurements overlap. This is also true for  $\sigma_y$  at 30 °C at the highest rate, see Fig. 5e. One possible explanation for this phenomenon is that the presence of micro-pores in the soft but not fully melted matrix can still act as weak links, reducing the yield and fracture stresses. However, the presence of more pores might not contribute significantly to any further stress reduction. This might be explained by the fact a portion of the chocolate matrix is melted whilst the rest still remains a solid. The pores in the melted part of the matrix might hinder the deformation and flow of the matrix. Therefore, competing mechanisms might occur at that semi-solid, semi-liquid state.

Despite the assumption that the computation of the macroscopic yield stress and Young's modulus might be subjective when the material exhibits such a non-linear stress-strain response, it is crucial to visualise the temperature dependence

of these material properties. To visualise this dependence and the drastic reduction of stresses with increasing temperature, the stress-strain response of the non-aerated chocolate at a strain rate of  $0.1 \text{ s}^{-1}$  for the three different testing temperatures is plotted in Fig. 6a. It becomes clear that stresses reduce due to the phase change (soft solid to viscous fluid) of the chocolate.

In addition, Fig. 6b summarises the data of the different chocolate materials as a function of temperature at a strain rate of  $0.1 \text{ s}^{-1}$ . Since the true modulus of elasticity needs to be computed from a loading-unloading experiment,<sup>22,33</sup> the values of the apparent modulus in the graph of Fig. 6b are normalised to the modulus of the given material at 20 °C. The graph is plotted on a logarithmic y-scale to distinguish the magnitude of the differences. Based on the same plot, the average values of Young's modulus are two orders of magnitude lower than the reported values at 20 °C, proving further the soft state of the chocolate at a temperature (*i.e.*, 30 °C) close to the phase change of the chocolate, which is reported to be 28–34 °C.<sup>8,36,37</sup>

Inspecting the fracture behaviour of the chocolate after the completion of the compression tests reveals differences in the





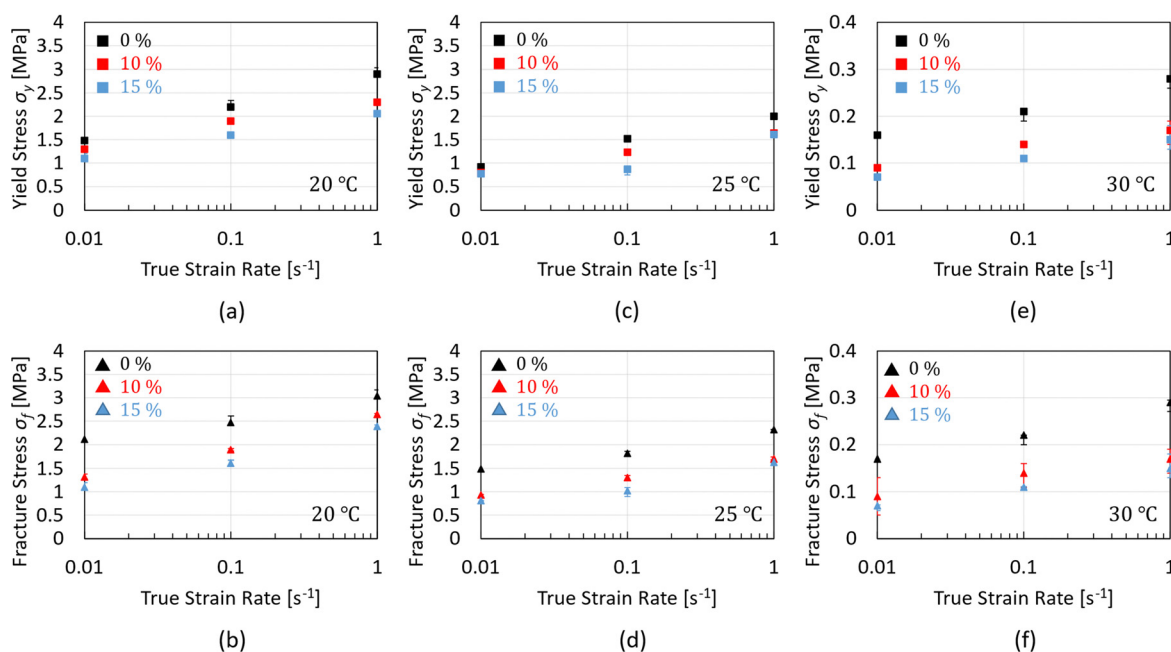


Fig. 5 Plots of the compressive macroscopic yield stress,  $\sigma_y$ , and fracture stress,  $\sigma_f$ , as a function of strain rate at 20 °C, 25 °C, and 30 °C for all chocolate materials.

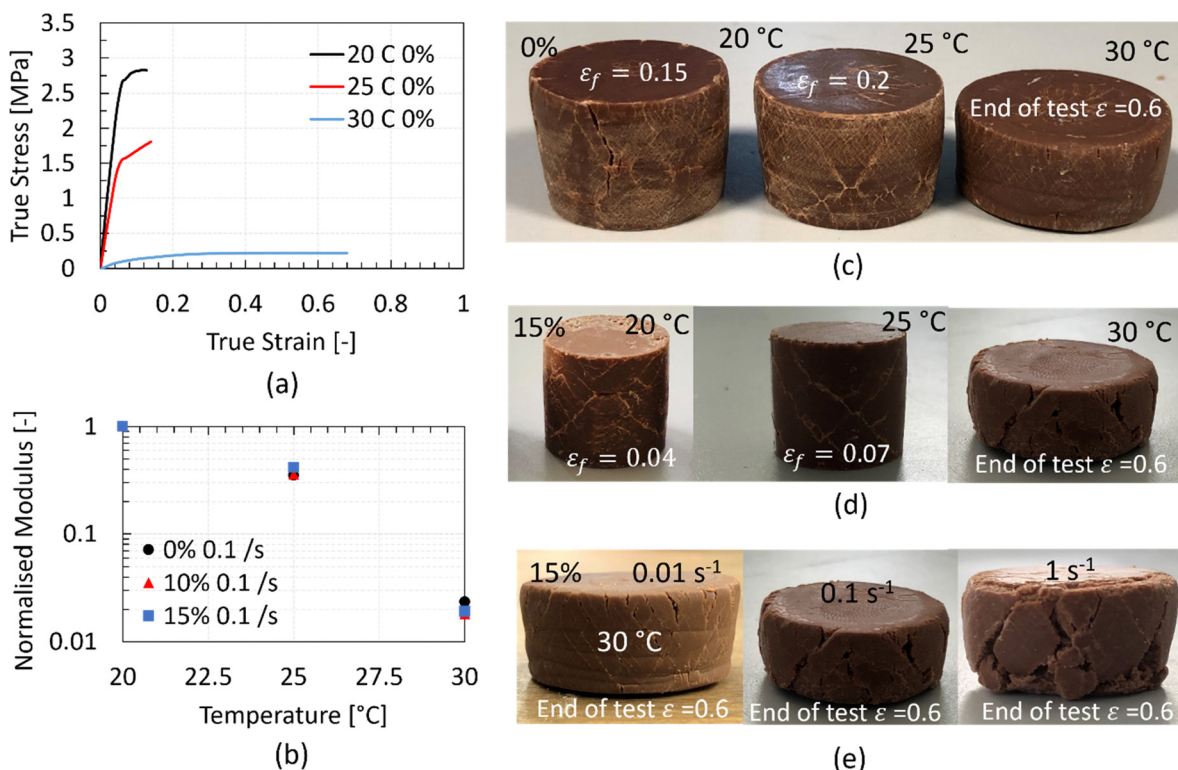


Fig. 6 (a) A typical stress–strain plot for the non-aerated chocolate under three different temperatures ( $\dot{\epsilon} = 0.1 s^{-1}$ ) obtained from the uniaxial compression tests; (b) normalised apparent elastic modulus to the modulus at 20 °C at a strain rate of  $0.1 s^{-1}$ , as computed from the initial linear part of the stress–strain curve, as a function of temperature. Fractured specimens at the end of the compression tests showing crack patterns of the (c) non-aerated and (d) 15% micro-aerated chocolate ( $\dot{\epsilon} = 0.01 s^{-1}$ ) under different temperature conditions; and (e) the 15% micro-aerated chocolate at 30 °C under three different strain rates. Note that the sub-images of figures (d) and (e) are not in scale.



shear band formation and the crack patterns for different testing temperatures and porosity levels. The presence of shear bands and the crack initiation and propagation along these bands implies a purely uniaxial compression behaviour for all chocolate samples.

The fractured specimens for the different cases are summarised in Fig. 6c–e. Fig. 6c and d summarise the fracture behaviour of the non-aerated and 15% micro-aerated chocolate materials, respectively, at different temperatures. For both chocolate materials, a decrease in the number of the shear band formation is observed with increasing temperature which is not surprising because the shear band formation is characteristic of brittle materials. The ductility of chocolate is found to increase with temperature (see larger fracture strains shown in Fig. 4 and 5d and f). In addition, no clear rate-dependent effect on shear band formation is reported for the 15% micro-aerated chocolate, see Fig. 6e, and similar effects are observed for the other two chocolate samples, *i.e.* 0% and 10% (results not shown). Interestingly, the cracks on the surface of the chocolate samples at 30 °C just after the test, as shown in Fig. 6e, imply that the chocolate exhibits characteristics of a solid, (*i.e.*, clear fracture due to the applied deformation) rather than a liquid (*i.e.*, flow due to deformation).

So far, the results from the uniaxial compression tests reveal a drastic dependence of the mechanical response of the non-aerated chocolate on the temperature. Mechanical properties such as the apparent Young's modulus, macroscopic yield stress, and compressive strength exhibit a reduction with increasing temperature. This trend persists even when comparing compression tests at 20 °C and 25 °C, which are both outside the phase change range, *i.e.*, (28–34 °C).<sup>8,36</sup> More specifically, the compressive stresses can decrease up to 30% when the temperature elevates by 5 °C, emphasizing the importance of maintaining controlled testing conditions, including temperature, humidity (its effect not studied here), and testing speed, in such complex biological systems like foods. The decrease in stresses and the further decrease in

chocolate's stiffness when the temperature rises by 10 °C illustrate the complexity of chocolate's mechanical response, transitioning from a “hard” solid at 20 °C to a very “viscous” fluid at 30 °C. These temperature-dependent mechanisms are also evident when considering the effect of temperature on the two micro-aerated chocolates, highlighting the soft nature of the chocolate at these temperature levels.

### 3.2. Rheological results

The viscosity–strain rate plots without and with the artificial saliva present, as presented in Fig. 7a and b, respectively, show that the viscosity decreases with increasing shear rate for all samples and increases with the increasing micro-aeration level. Fig. 7a shows the comparison among the viscosity values without artificial saliva for samples with porosities  $f = 0, 10$ , and 15 vol%. The trend for all samples is the same, the viscosity is initially steady for low shear rates and reduces for shear rates between 0.2 and 600  $\text{s}^{-1}$  as the shear rate increases until it reaches a constant value at shear rates above  $\dot{\gamma} > 600 \text{ s}^{-1}$ .

Fig. 7b shows results of viscosity for the different micro-aeration levels with the artificial saliva present. The same trend is observed, although there is no difference in viscosity values, for shear rates above  $\dot{\gamma} > 600 \text{ s}^{-1}$ . Clear differences were observed in viscosity values for all micro-aeration levels with and without artificial saliva present in both figures. The Carreau and Cross models were fitted to the experimental results and both of them fit well as shown in Fig. 7a and b. Tables 1 and 2 summarise the calibrated parameters of the two models without and with artificial saliva, respectively. The parameter  $p$  has a constant value of 0.72 whereas the parameter  $m$  ranges between 1.48 and 1.5. The parameter  $p$  is higher than the corresponding  $p$  value of 0.42 reported by Taylor *et al.*,<sup>50</sup> who examined the rheology of molten crumb chocolate. The constant  $\lambda_c$  appears to be higher than those reported in the literature for non-aerated chocolate samples, 2.9 *vs.* 1.9.<sup>50</sup> The results showed that the rheological properties of the examined chocolate samples can be modelled well using both the Carreau and Cross models.

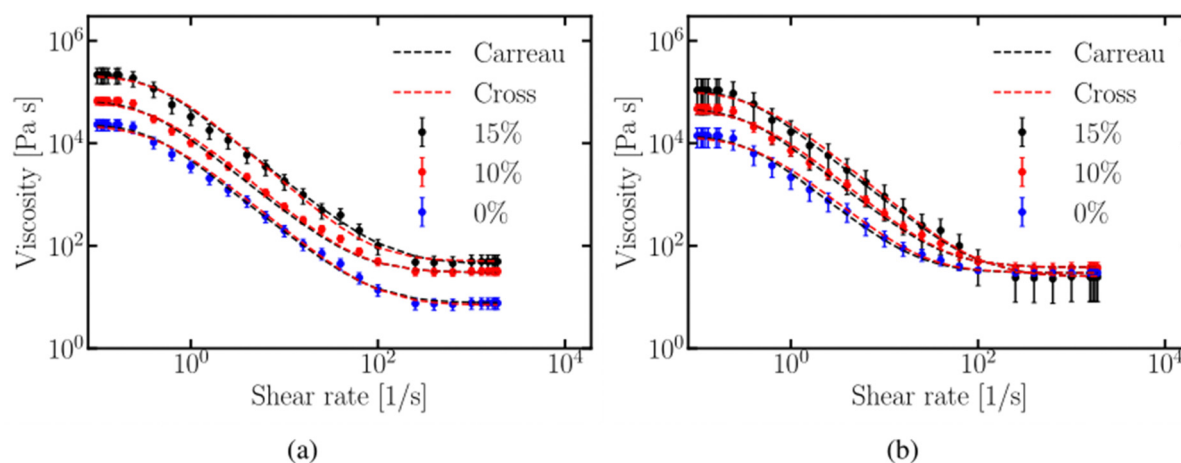


Fig. 7 Flow behaviour of the three chocolate samples with different micro-aeration levels (a) without and (b) with artificial saliva present at 37 °C. The calibrated Carreau and Cross models are also depicted in both figures.



**Table 1** Carreau and Cross calibrated parameters for different micro-aeration levels without artificial saliva present

Parameter	$f = 15 \text{ vol}\%$	$f = 10 \text{ vol}\%$	$f = 0 \text{ vol}\%$
$\eta_0$ [Pa s]	218 000	67 191	23 373
$\eta_\infty$ [Pa s]	44	30	7.71
$\lambda_c$ [s]	2.7	3.2	2.9
$\alpha_c$ [s]	2.1	2.5	2.4
$p$ [—]	0.72	0.72	0.72
$m$ [—]	1.6	1.5	1.48

**Table 2** Carreau and Cross calibrated parameters for different micro-aeration levels with artificial saliva present

Parameter	$f = 15 \text{ vol}\%$	$f = 10 \text{ vol}\%$	$f = 0 \text{ vol}\%$
$\eta_0$ [Pa s]	109 000	47 048	14 049
$\eta_\infty$ [Pa s]	25	38	29.5
$\lambda_c$ [s]	3.1	3.2	3.1
$\alpha_c$ [s]	2.4	2.5	2.4
$p$ [—]	0.72	0.72	0.72
$m$ [—]	1.6	1.5	1.48

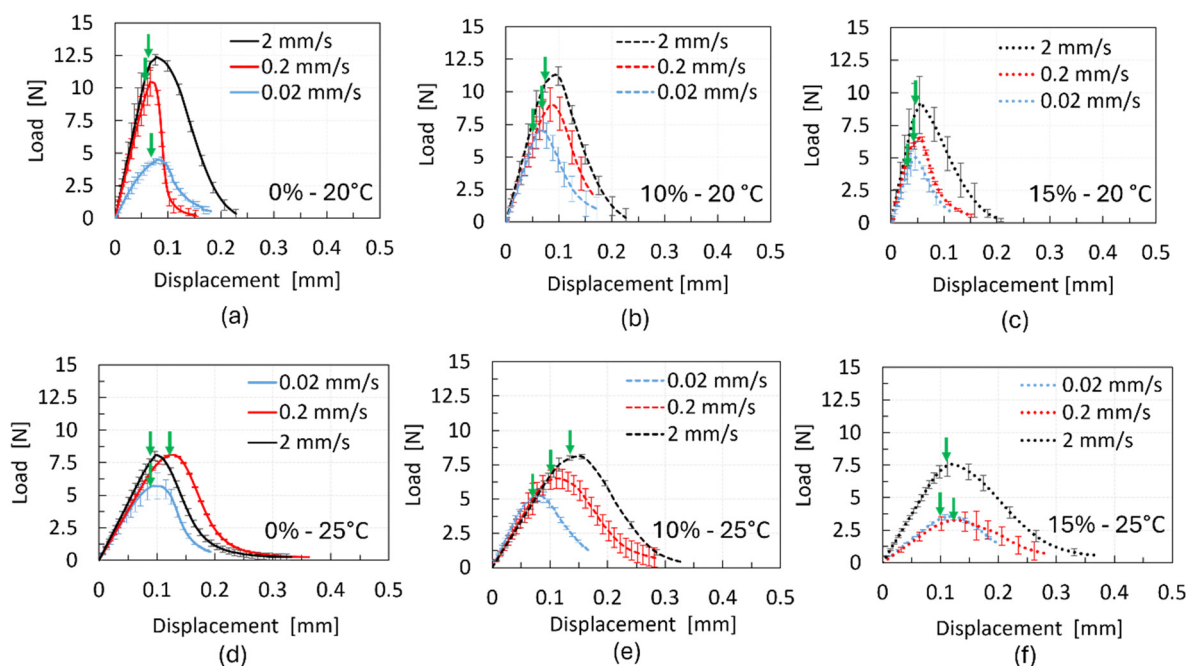
Micro-aeration increases the viscosity of chocolate by modifying its rheology, leading to greater shear-thinning and viscoelastic behaviour. Studies on aerated liquids, like honey and syrup, have shown that air bubbles resist deformation at low shear rates, increasing relative viscosity. Torres *et al.*<sup>51</sup> found that pores in the chocolate matrix influence shear thinning. At low shear rates, micro-pores act as barriers to deformation, increasing viscosity, while at high shear rates, they align with the flow, causing shear thinning. The capillary number ( $Ca$ ) describes micro-void deformation and is calculated as:

$$Ca = \frac{\eta_0 r \dot{\gamma}}{\Gamma}$$

where  $\eta_0$  is the viscosity of the liquid,  $r$  is the radius of the undeformed spherical pores,  $\dot{\gamma}$  is the shear rate and  $\Gamma$  is the surface tension at the pore-liquid interface. The equilibrium deformation is determined by the balance between the deforming, viscous stress  $\eta_0 \dot{\gamma}$  acting on a pore and the restoring stress supplied by surface tension ( $\Gamma/r$ ). The shear thinning behaviour of aerated liquids has been reported in the literature initially in the study by Shaw *et al.*<sup>52</sup> who investigated the effect of bubbles in magma viscosity and the later study of Manga *et al.*<sup>53</sup> validated his findings. The capillary number has been related to the magnitude of pore deformation  $D = (A - B)/(A + B)$ , where  $A$  and  $B$  are the major and minor axes of the deformed pore, respectively.<sup>53</sup> Assuming that the pores in the chocolate matrix are initially undeformed ( $D = 0$ ), the deformation induced by shear will increase  $D$  resulting in higher  $Ca$ . The flow streamlines are more distorted in the case of the spherical pore which act as a barrier to the flow, resulting in higher viscosity values. The viscosity reduction of micro-aerated chocolate follows the same trend as the bubble-bearing magmas studied by Rust and Manga.<sup>54</sup> This theory can also be used to explain the similar viscosity values obtained for high strain rates when saliva is present in Fig. 7b. The addition of saliva possibly enhances the flow of the melted chocolate around the deformed pores, reducing the effect that micro-aeration has on the viscosity.

### 3.3. Fracture results

Fig. 8 shows the typical load-displacement response for all three chocolate materials under three testing speeds of  $0.02 \text{ mm s}^{-1}$ ,  $0.2 \text{ mm s}^{-1}$ , and  $2 \text{ mm s}^{-1}$  at  $20^\circ\text{C}$  and at  $25^\circ\text{C}$ . The larger error bars compared to the uniaxial compression tests are attributed to the challenge of precisely controlling the length and sharpness of the notch due to the brittle nature

**Fig. 8** Load-displacement data for chocolate materials of different porosities from the SENB tests under three testing speeds at  $20^\circ\text{C}$  and  $25^\circ\text{C}$ .

of the material. Based on Fig. 8, the load increases nonlinearly, reaching a maximum load followed by a smooth drop to zero load, implying complete fracture of the surfaces. In all cases, the maximum load increases with increasing testing speed, increasing the total area underneath the force–displacement curve. The gradual drop of the load suggests a stable crack growth for all three chocolate materials which progresses almost asymptotically to the zero value (see Fig. 8). The arrows in the plots of Fig. 8 indicate the crack initiation of all tested materials for each testing speed, which is found to be close the maximum points, as observed from video images recorded during the experiments. After the completion of the experiments, separation of the cracked surfaces was not observed regardless of the testing speed and chocolate material. The cracked surfaces remained in contact, implying the presence of an adhesion mechanism that holds the two surfaces together even when the samples returned to the zero-force position.

The critical strain energy release rate,  $G_c$ , of the three chocolate varieties is summarised in Fig. 9 as a function of crack speed. Based on the plots of this figure, increasing trends of  $G_c$  with increasing crack speed are observed for all materials at both testing temperatures. Based on Fig. 9a, the  $G_c$  values of the 0%, 10% and 15% micro-aerated chocolate at 20 °C range from 8–12 J m<sup>-2</sup>, 5.5–10 J m<sup>-2</sup> and 4–8 J m<sup>-2</sup>, respectively. The same ranges at 25 °C change to 6–15 J m<sup>-2</sup>, 5–13 J m<sup>-2</sup> and 5.2–13 J m<sup>-2</sup>, respectively. The decreasing trend of  $G_c$  with increasing micro-aeration is evident at both testing temperatures. Based on these data, it is argued that the micro-pores dispersed in the chocolate microstructure impact  $G_c$  of the porous chocolate, enhancing the material's brittle nature, which is aligned with reports in the literature for porous or cellular materials.<sup>55,56</sup> The  $G_c$  values of chocolate shown in Fig. 9 are found to be somewhat different to  $G_c$  of chocolate ( $\sim 2.1$  J m<sup>-2</sup>) reported by Gomez *et al.*,<sup>20</sup> who inversely calculated the critical strain energy release rate of chocolate using a finite element model. This discrepancy potentially implies variability in the material properties of chocolate depending on its composition or even highlights the need for rigorous mechanical analysis to get an accurate value of fracture toughness.

Moving to the wire cutting experimental results, Fig. 10 summarises the cutting force *versus* wire displacement results for all testing conditions and chocolate varieties. Based on Fig. 10, the cutting force increases with growing wire diameter, both in the indentation and the steady state cutting regime. This trend aligns with the analysis presented in Section 2.3.3 assuming that the total energy increases with increasing wire diameter due to other forms of energy present during the cutting process, such as friction energy and plastic deformation local to the crack tip, and not the fracture energy. The macroscopic behaviour of the chocolate during the wire-cutting experiments exhibited characteristics of a solid, *i.e.*, clear cracks forming in front of the wire rather than flowing, as shown in images of Fig. 2d. This observation is consistent with the experimental results from the uniaxial compression tests at 30 °C, where clear cracks and fractures were observed (see Fig. 6e).

For a given wire diameter (see Fig. 10 for a given type of line), the peak force in the cutting force–wire displacement curves does not change with increasing testing speed (shown in Fig. 10 with different colours). Any differences observed among different speeds for the cases with the larger wire diameter of 0.5 might be due to the state of the material, which is only partially melted. The fact that the region in front of the wire might be melted or not in some cases might contribute to these differences and create a variability in the cutting force–wire displacement data. However, focusing on the steady-state regime of the curve (see Fig. 3b), which regards the critical data for the calculation of  $G_c$ , rate-dependent effects might be more prominent.

Fig. 11 shows the surface cuts under three wire diameters of 0.5 mm, 0.25 mm, and 0.125 mm for the case of the non-aerated and the 15% micro-aerated chocolate. Note that the surface cuts between the 10% and 15% micro-aerated chocolate samples did not exhibit any distinct differences. In both figures, there are two distinct regions in the fracture surface of the chocolate, as shown in Fig. 11, denoted as “i” and “ii”. The “i” region corresponds to the surface cut by the wire and the second, *i.e.*, “ii”, indicates the natural fracture surface

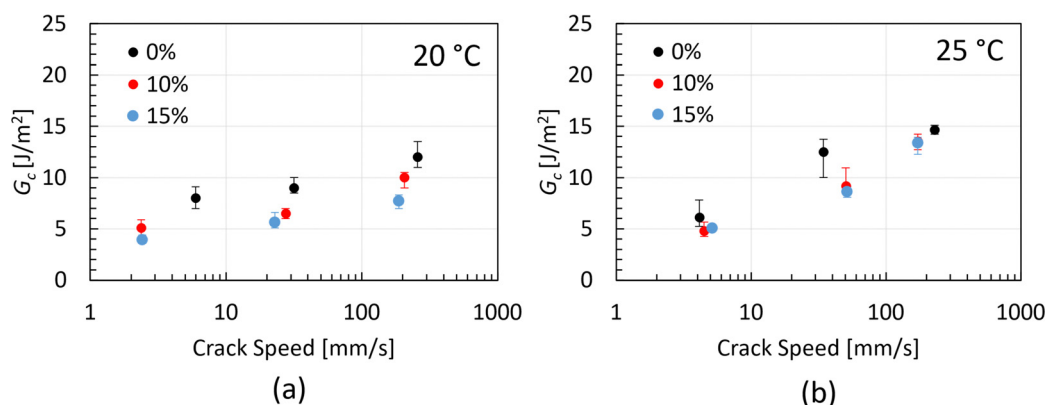


Fig. 9 Critical strain energy release rate data obtained from the SENB experiments for all types of chocolates at 20 °C and 25 °C as a function of the crack speed.



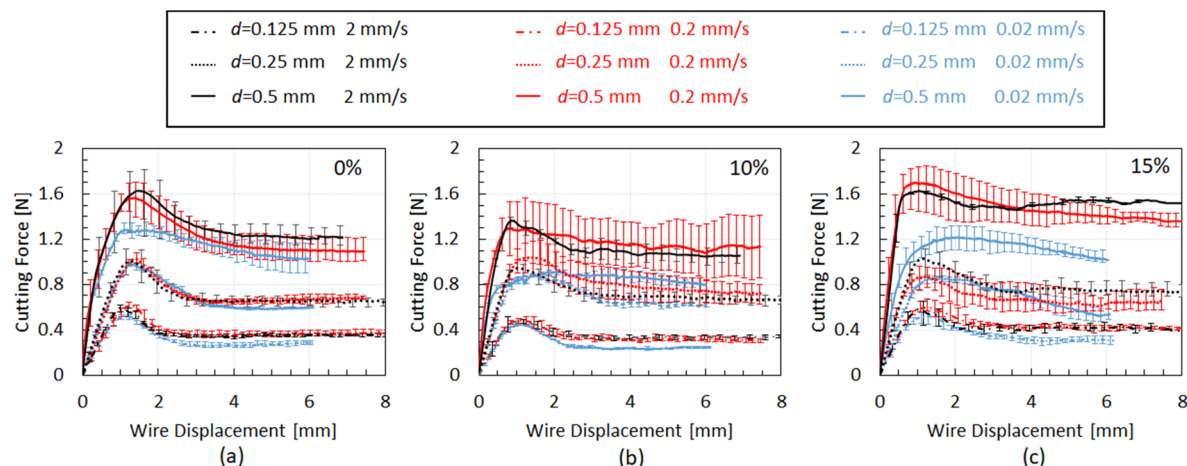


Fig. 10 Typical cutting force–wire displacement plots for the (a) non-aerated, (b) 10% micro-aerated, and (c) 15% micro-aerated chocolate for different wire diameters and crack speeds.

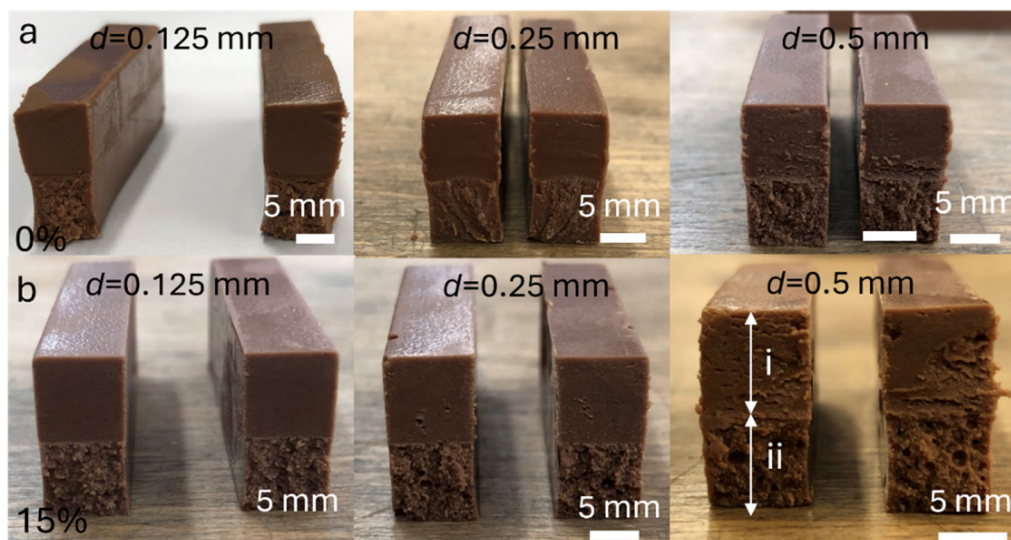


Fig. 11 Images showing the fracture surface of the (a) non-aerated and (b) the 15% micro-aerated chocolate collected at the end of the wire cutting experiments. The images show the fracture surface of the two types of chocolate at different wire diameters at a constant crack speed of  $0.2 \text{ mm s}^{-1}$ . The distances “i” and “ii” depict the deformed surface by the wire and the surface which is naturally cracked, for inspection, at the end of the test. The direction of the cuts was from top to bottom.

produced after the experiment to inspect the fracture surfaces. The fractured surface “i” indicates a larger degree of damage with increasing wire diameter. In addition, in the surface cuts produced by the wire of 0.5 mm diameter, a tearing effect in the cracked surface is depicted for all types of chocolate. This tearing effect at the cracked surfaces is caused by secondary cracks occurring during the wire cutting experiments at large sizes of wire diameter.<sup>2,45</sup> This tearing effect diminishes with decreasing wire diameter to produce a smooth, flat surface, at least macroscopically, at  $d = 0.125 \text{ mm}$ , which was true for all tested materials. The surface cuts shown in Fig. 11 and this tearing effect are consistent with other studies on foods such as cheese and gels.<sup>2,45</sup>

Comparing the surface cuts between the two chocolate materials, the tearing effect in the case of the 15% micro-aerated

chocolate occurs more predominately compared to its solid 0% counterpart. It is evident that micro-aeration affects the damage process that occurs during the wire cutting experiments which potentially correlates with the larger error bars present in cutting force–wire displacement data of these materials, as shown in Fig. 10. This mechanism is expected to have an impact on  $G_c$ .

The cutting data in the form of  $\frac{F}{B} \left( \frac{\dot{\epsilon}_0(d+C)}{K\dot{a}} \right)^{m_w}$  vs.  $d \left( \frac{\dot{\epsilon}_0(d+C)}{K\dot{a}} \right)^{m_w - n_w}$  plots are shown in Fig. 12. The  $G_c$  values of the different materials, which correspond to  $d = 0 \text{ mm}$ , are shown in the equation of the trend lines (constant term of the equation) and summarised in Fig. 12d for different testing speeds. Considering the relatively large error bars in the bar chart, no clear rate effect on  $G_c$  can be suggested for any of the



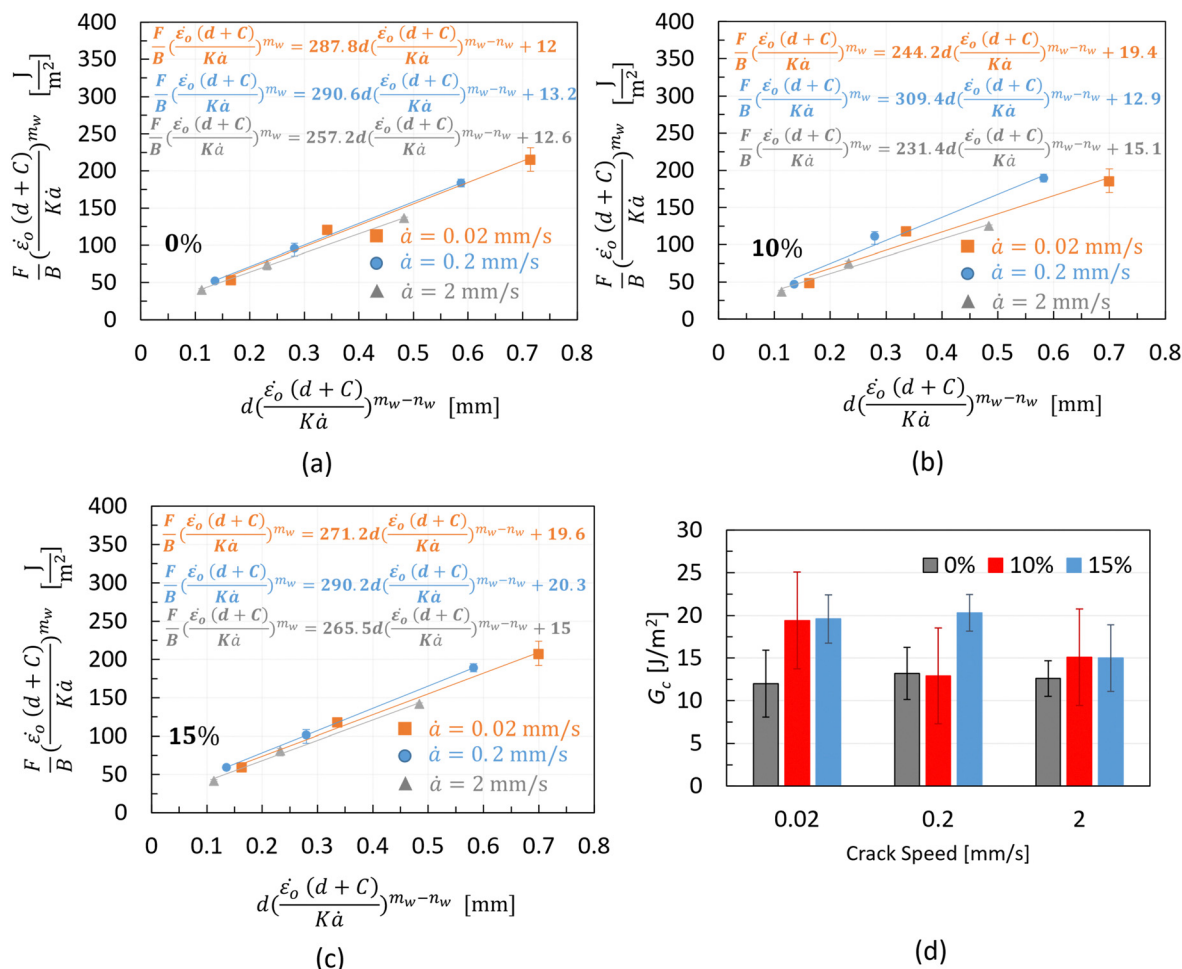


Fig. 12 Cutting force over sample's thickness data, using eqn (15), along with the fitted trend lines for three testing speeds and the (a) 0%, (b) 10%, and (c) 15% chocolate. Bar chart (d) summarising the critical strain energy release rate values extracted from the wire cutting tests through the extrapolation method for all types of chocolate, 0%, 10%, and 15%, as a function of the cutting speed.

materials. These error bars, reaching up to 50% of the average value in some cases (e.g., non-aerated chocolate at the slowest cutting speed in Fig. 12d), result from the extrapolation process to derive  $G_c$ . Even a small variation amongst the repeats, e.g. 3% for a certain wire diameter, could produce significant errors in the average  $G_c$ , exceeding 10–15%. Similar magnitudes of error bars are observed in other studies employing the wire cutting technique.<sup>45</sup>

The results shown in Fig. 12d do not indicate a clear trend in the impact of micro-aeration on  $G_c$ . In addition, at the lowest and highest testing speeds, there is no difference between  $G_c$  of the 10% and 15% micro-aerated chocolate. However, at the intermediate speed of  $0.2 \text{ mm s}^{-1}$ , there is a large difference between the two chocolates. This inconsistency might be due to the soft, partially melted, state of the chocolate at this temperature.

Unlike the compression tests, which measure the bulk response of the chocolate sample and the SENB test, where the indenter exhibits significantly larger than the wire (radius: 4 mm vs. 0.25 mm), the wire-cutting experiments test a small region ahead of the wire. The difference in microstructure

between the 10% and 15% chocolate in a plane with such dimensions (Fig. 1b and c) might not be sufficient to produce consistent differences in  $G_c$ , as opposed to the differences reported from the SENB tests and compression tests.

However, the bar chart results indicate some increase in  $G_c$  for the micro-aerated material compared to the non-aerated product, in contrast to the observations from the SENB tests at  $20^\circ\text{C}$  and  $25^\circ\text{C}$ . The same chart shows that the difference in  $G_c$  between non-aerated and micro-aerated chocolate is more distinct at the slowest speed compared to the highest testing speed. This suggests a different fracture behaviour for the micro-aerated chocolate depending on the testing rate, consistent with reports in the literature for other multiphase materials.<sup>57,58</sup> At slow speeds, fracture occurs in regions with imperfections, micro-cracks, pores, or imperfect interfaces, while at higher speeds, fracture involves a higher stress state that can directly crack the matrix. This argument is believed to explain the results in Fig. 12d for the micro-aerated chocolate.

Assessing the dependency on temperature, as obtained by the different fracture experiments,  $G_c$  is plotted against crack speed for all temperature thresholds and materials, and the



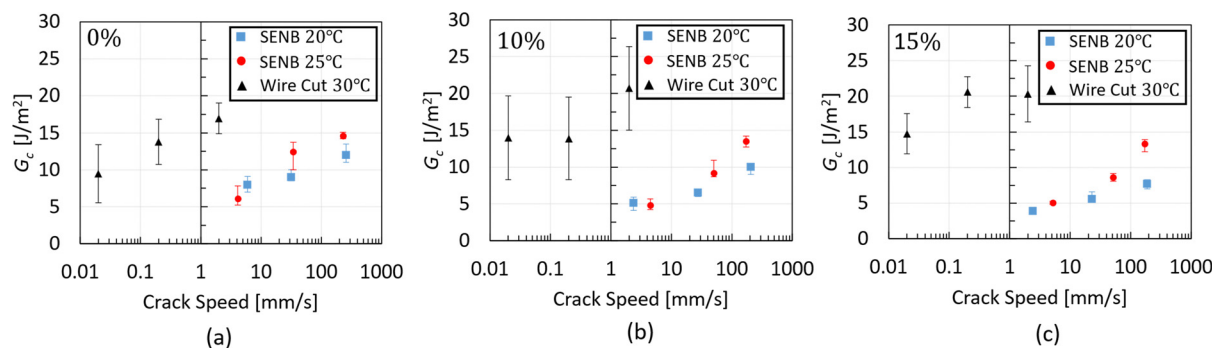


Fig. 13 Graphs summarising  $G_c$  obtained from SENB and wire cutting tests at different temperatures, 20 °C, 25 °C, and 30 °C, as a function of crack speed. Plots show the average values of  $G_c$  of the (a) non-aerated chocolate (0%), (b) 10%, and (c) 15% micro-aerated chocolate.

results are summarised in Fig. 13a–c for the 0%, 10%, and 15% micro-aerated chocolate, respectively. For a valid comparison between the SENB and wire cutting tests,  $G_c$  values that fall within the same crack speed region for all testing conditions are only compared. These regions correspond to 1–4 mm s<sup>−1</sup>, for the comparison of the  $G_c$  data at 20 °C and 30 °C, and 180–250 mm s<sup>−1</sup>, for the comparison of the SENB  $G_c$  data at 20 °C and 25 °C.

For all chocolate materials, a significant increase in  $G_c$  with temperature is observed. This is evident by comparing the SENB  $G_c$  values at 20 °C with the corresponding  $G_c$  values at 25 °C at the highest speeds (180–250 mm s<sup>−1</sup>), and similarly for the SENB  $G_c$  at 20 °C and the wire cutting  $G_c$  at 30 °C at speeds between 1 mm s<sup>−1</sup> and 4 mm s<sup>−1</sup>. At the highest speeds, the difference between SENB  $G_c$  values at 20 °C and 25 °C for the 0%, 10%, and 15% micro-aerated chocolate is approximately 22%, 35%, and 72%, respectively. On the other hand, for the speed range of 1–4 mm s<sup>−1</sup>, a difference of 60% is observed between SENB  $G_c$  at 20 °C and the wire cutting  $G_c$  at 30 °C for the non-aerated chocolate.

This difference increases to 75% and 78% for the 10% and 15% micro-aerated, respectively. The increased “jump” in the  $G_c$  value at 30 °C of the micro-aerated materials compared to their solid counterpart may be attributed to the presence of the micro-pores in the chocolate matrix. During the wire-cutting test, the presence of micro-pores dispersed in the chocolate matrix might impact the fracture process by creating a larger degree of secondary cracks and a tortuous crack path in front of the wire tip, thereby increasing  $G_c$  of the micro-aerated chocolate at 30 °C. This hypothesis is supported by the observation of a greater degree of damage observed in the fracture surfaces of the micro-aerated chocolate (resulting in a “rougher” surface) compared to the non-aerated counterpart, as shown in Fig. 11a and b, as well as by the larger cutting forces observed for the micro-aerated material in Fig. 10. This hypothesis is also supported by the rheological results at 37 °C in this work, which demonstrate an increase of the overall viscosity with increasing level of micro-aeration, see Fig. 7. Based on these observations, the presence of micro-pores in the chocolate system seems to obstruct the flow. A similar effect might also be present at lower temperatures close to 30 °C during the wire cutting experiments.

It is important to highlight that at this temperature, the material exhibits characteristics of both a viscous liquid and a soft solid, questioning the conventional definition of  $G_c$ . More specifically, there is a dual argument: one could argue that the material fractures by the action of the wire, creating a crack tip in front of the wire, and concurrently, it may also undergo flow around the wire during the fracture experiments. This dual nature argument was initially introduced by Skamniotis *et al.*,<sup>59</sup> who investigated the fracture behaviour of oatcake biscuits with varying moisture levels. This effect might also be relevant to chocolate at temperatures close to the phase change temperature, *i.e.*, 28–34 °C, where the material becomes softer and potentially  $G_c$  measurements increase because the energy dissipation due to deformation ahead of the crack tip becomes significantly higher than the actual energy due to fracture.

Investigating the impact of micro-aeration on mechanical properties at 20 °C reveals a reduction in fracture stresses, strains, and  $G_c$  with increasing micro-aeration, consistent with trends observed in engineering materials (pores create weaker structures). However, at 30 °C, the trends begin to shift: (i) fracture toughness starts to increase with micro-aeration, attributed to micro-pores hindering the fracture process, and (ii) fracture stress and fracture toughness do not change significantly with a further 5% increase in micro-aeration. In addition, the viscosity measurements at 37 °C showed that micro-pores hinder the flow process, leading to increasing trends of viscosity with micro-aeration.

The results of these sections highlight how the impact of micro-aeration changes as the testing temperature increases and the material transitions from a soft solid to a viscous liquid. The discussed trends suggest that structural changes in chocolate due to temperature (and partial or complete melting) might induce microstructural movements or mechanisms that affect the material behaviour; micro-aeration, even at the small levels of 10–15%, can in turn significantly affect these mechanisms.

## 4. Discussion on micro-aeration and its link to consumer perception

This investigation of chocolate proposes a shift from the classical solid and fracture mechanics theory when it is considered as a





solid, towards the theory of rheology when the material is considered fluid. The authors suggest that for applications where temperature variations lead to a phase change of chocolate coupling of classical solid and fracture mechanics experiments with rheological experiments could provide a better understanding of the material's behaviour.

An example of such a scenario is found in the context of food oral processing, where a progressive structural breakdown of chocolate occurs due to mechanical, thermal, and enzymatic loads.<sup>27,28</sup> From our previous published work,<sup>27</sup> temporal dominance sensation (TDS) tests offered valuable insights into the processes and sensations perceived during the chocolate's oral processing. A modified version of the TDS graph from that study is presented in Fig. 14 for two chocolate materials with different levels of porosity, 0% and 10%. In these plots, only textural attributes which are statistically dominant, *i.e.* lying above the significance line (see red dotted line) are shown for both cases. The onset of food oral processing (FOP) is assumed when the first sensation becomes dominant, and swallowing (end of FOP) occurs when the mouthcoating sensation is perceived.

Based on the results of our previous published work, which focused solely on mechanical measurements at 20 °C, the increased dominance (peak value of the light-blue curve in Fig. 14) in the “soft” perception between the non-aerated and 10% micro-aerated chocolate is justified by the reduction of Young's modulus and fracture stresses at 20 °C.<sup>27</sup> Furthermore, the increase in fragmentation, as illustrated in that study by the *in vivo* mastication tests, with 10% micro-aeration, is linked to the decreased  $G_c$  from SENB tests at 20 °C. This increase in fragmentation is also thought to accelerate the melting, thereby increasing dominance and the duration of “smooth” perception (see the green curve in Fig. 14). Lastly, the absence of the sticky attribute as the dominant sensation in the case of the 10% micro-aerated chocolate was attributed to the drop in friction with increasing micro-aeration.

In addition to the above arguments, the new combined engineering approach of this study offers insights into the trends and evolution of certain sensorial attributes. For instance, based on Fig. 14b, the prolonged “soft” sensation in the case of the micro-aerated chocolate is attributed to a synergy of the solid-like behaviour during the early stages of FOP (time: ~10–15%) and the complex liquid behaviour of the material during later stages when melting is expected to initiate (time: ~30–40%). The reduced fracture stresses of the micro-aerated chocolate compared to its solid counterpart justify the absence of a “hard” sensation (and the first peak of a “soft” sensation) during FOP times between 10 and 15% as the chocolate remains predominantly solid at this stage. Conversely, the delayed peak of the “soft” sensation curve (FOP time: 30–40%) is believed to be linked to the increased  $G_c$  at 30 °C and its flow behaviour at 37 °C, indicating the onset of melting (as implied by the presence of “moist” sensation). The wire cutting experiments at 30 °C revealed higher  $G_c$  values with micro-aeration (see Fig. 13) and potentially higher energy dissipation mechanisms due to the larger energy to create a new crack, as well as the presence of secondary cracks

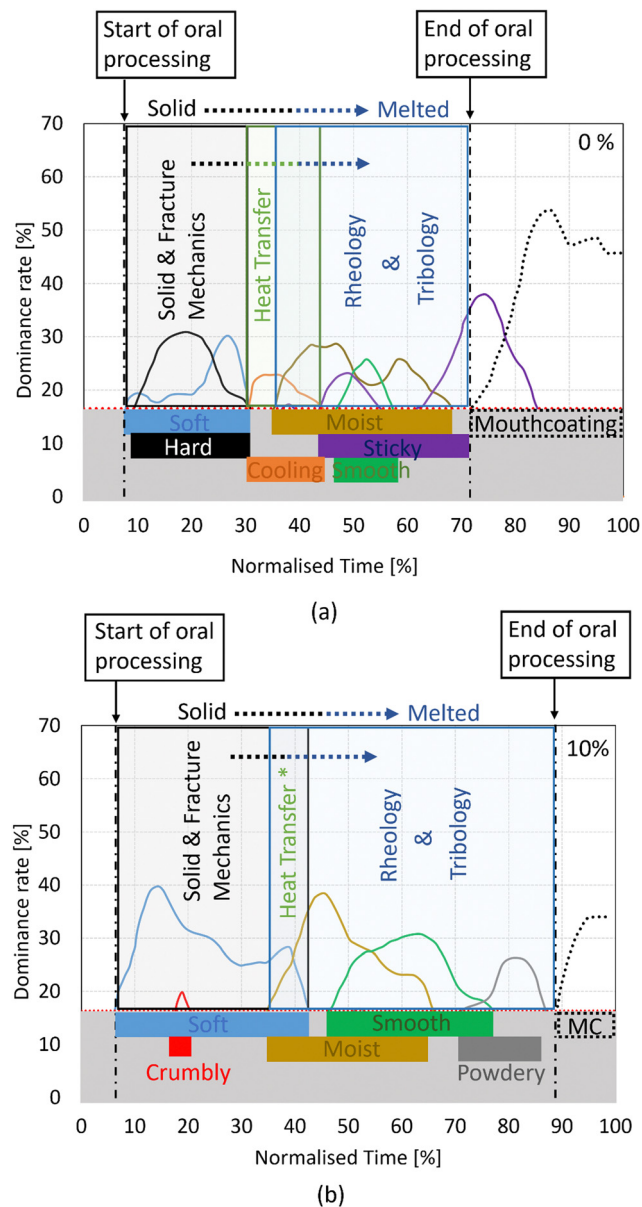


Fig. 14 Temporal dominance sensation (TDS) sensory results in the form of a graph for (a) non-aerated (0%) and (b) 10% micro-aerated chocolate. The graph was adapted from Bikos *et al.*<sup>27</sup> and only textural results which are statistically significant, *i.e.*, above the red dotted line, are shown in this graph. The start and end of oral processing are assumed when the first sensation and once the mouth-coating is perceived. Mouthcoating (or “MC” in image b) is a sensation which is assessed after the material is consumed.<sup>27</sup>

(see Section 3.3). The overall increased viscosity with micro-aeration (see Fig. 7) could also create a resistant viscous chewed mass (partially melted) which could contribute to a prolonged “soft” sensation.

The reduction of  $G_c$  with micro-aeration at temperatures of 20 °C and 25 °C (see Fig. 13), together with the reduction of the fracture strains (see the change in the lateral position of the red marks in Fig. 4) with micro-aeration, implies a more brittle behaviour for the micro-aerated chocolate. This suggests higher





fragmentation of the micro-aerated material, leading to an expedited melting process, as reported in previous literature.<sup>27</sup> The expedited melting process could affect the perception of attributes associated with the melting of chocolate,<sup>60</sup> such as the “cooling” sensation in this study. In this case, an accelerated melting process might diminish the perceived prominence of the “cooling” sensation compared to other attributes sensed at the same time.

Finally, the increase in the viscosity measurements with micro-aeration at shear rates relevant to FOP (1–20 Hz) results in a more viscous bolus, which may be more coherent and tolerant to the shear actions during the subsequent steps of the FOP, contributing in that way to the increased dominance and duration of the “smooth” perception in the 10% micro-aerated chocolate.

The definition of the textural attributes can reveal the state of the chocolate during food oral processing (FOP).<sup>27</sup> For instance, “hardness–softness” and “crumbly” attributes are associated with the solid nature of the chocolate and are perceived in the early stages of FOP, while “cooling” indicates the melting of chocolate and its phase change. Attributes such as “smooth–powdery–gritty”, “dry–moist”, and “sticky” are related to the molten state of chocolate and the bolus formation.

Based on this argument, the authors propose the temporal dominance of engineering analysis including solid and fracture mechanics, heat transfer, and rheology and tribology, each relevant to specific stages of the FOP. The duration of each dominant engineering approach – solid and fracture mechanics, heat transfer, and rheology and tribology – is aligned with the dominance of the corresponding textural attribute. For instance, the regime where the solid and fracture mechanics analysis is dominant aligns with the onset and the end of the “hard/soft” sensation. For heat transfer, the regime coincides with the duration of the “cooling” sensation. Lastly, rheology and tribology<sup>27,28</sup> are important to consider when attributes such as “moist” and “sticky” become dominant. These dominant regimes are depicted in the plots of Fig. 14. It is important to note that the absence of the cooling attribute in Fig. 14b does not negate the importance of heat transfer analysis, as chocolate melting is associated with not only textural but also flavour attributes.<sup>61</sup>

It is important to highlight that these arguments represent the authors' interpretation based on their rigorous engineering analysis. However, extracting valid correlations between the material properties and the sensory perceptions remains a challenge that requires further investigation.

The material properties defined in this work using various experimental methods as a function of rate, temperature, and level of micro-aeration present an opportunity for advancing numerical models in the literature. One example is the advancement of recent multiscale models that simulate the first bite of chocolate<sup>3,34</sup> to incorporate mastication and account for the effect of temperature as the material transitions from solid to liquid. Such computational tools can leverage the mechanical properties established in this study, considering their dependence on temperature and rate, to unravel the mechanisms occurring in these intermediate stages of oral

processing. The advantage of such computational models over *in vivo* or *in vitro* experiments is that simulating consumer-specific testing conditions such as chewing rate, chewing force and their effect on the breakdown of chocolate is a feasible task. This paper lays the foundation by presenting a detailed methodology for accurately estimating the material properties of chocolate, essential for developing such a computational model.

## 5. Conclusions

Our study provides a comprehensive experimental methodology that combines fracture and rheological theories to measure the rheological and fracture properties of chocolate with varying micro-aeration levels as a function of rate and temperature. The combined experimental campaign reveals a drastic dependence of the material properties of chocolate on micro-aeration. Specifically, fracture stresses and the critical strain energy release rate,  $G_c$ , decrease with micro-aeration when the chocolate is solid, resulting in a more brittle and soft material. However, at 30 °C,  $G_c$  is found to increase with micro-aeration due to the obstruction of the fracture process by the micro-pores. Similarly, at 37 °C, the rheological experiments show that the presence of micro-pores leads to a more viscous flow, likely caused by the increased capillary number. These results demonstrate how the microstructure impacts the material properties of chocolate differently during the transition from a solid to a viscous liquid. The relationships between the micro-aeration and material properties and the dependence of temperature on the different mechanical properties as defined by the combined engineering analysis are used to explain the difference in textural attributes observed in micro-aeration during temporal dominance sensation tests. Moreover, these material parameters serve as valuable inputs for computational models that simulate the fracture and the phase change of chocolate during oral processing.

## Author contributions

G. Samaras and D. Bikos: conception of the idea, investigation, methodology, formal analysis, writing the original draft, writing, review, and editing of the revised manuscript. M. N. Charalambides: conception of the idea, writing the original draft, writing, review, and editing of the revised manuscript, project administration, supervision, funding acquisition. P. Cann, M. Masen, Y. Hardalupas, J. Vieira, C. Hartmann: validation, review and editing the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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