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# Wide-field imaging of active site distribution on semiconducting transition metal dichalcogenide nanosheets in electrocatalytic and photoelectrocatalytic processes†

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Semiconducting transition metal dichalcogenide (TMD) nanosheets are promising materials for electrocatalysis and photoelectrocatalysis. However, the existing analytical approaches are inadequate at comprehensively describing the operation of narrow-bandgap semiconductors in these two processes. Furthermore, the distribution of the reactive sites on the electrode surface and the dynamic movement of carriers within these semiconductors during the reactions remain ambiguous. To plug these knowledge gaps, an *in situ* widefield imaging technique was devised in this study to investigate the electron distribution in different types of TMDs; notably, the method permits high-spatiotemporal-resolution analyses of electron-induced metal-ion reduction reactions in both electrocatalysis and photoelectrocatalysis. The findings revealed a unique complementary distribution of the active sites on WSe<sub>2</sub> nanosheets during the two different cathodic processes. Our facile imaging approach can provide insightful information on the heterogeneous structure–property relationship at the electrochemical interfaces, facilitating the rational design of high-performance electrocatalytic/photoelectrocatalytic materials.

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## Introduction

Two-dimensional (2D) nanosheets of transition metal dichalcogenides (TMDs) such as WSe<sub>2</sub> and MoS<sub>2</sub> exhibit noteworthy electrocatalytic and photoelectrocatalytic activities in various important reactions such as water splitting,<sup>1</sup> N<sub>2</sub> reduction,<sup>2</sup> CO<sub>2</sub> reduction.<sup>3</sup> Notably, the catalytic performance of these 2D materials at electrochemical interfaces relies heavily on their local atomic structures, *e.g.*, defect sites, edges, and corners;<sup>4–6</sup> however, these units can serve as detrimental recombination centers<sup>7</sup> or beneficial catalytic sites.<sup>8</sup> Therefore, these active sites must be investigated using *in situ* characterization techniques to comprehensively understand the dynamic structure–activity relationship of catalysts during catalytic reactions,<sup>9–11</sup> thereby improving catalytic performance.<sup>12</sup>

In terms of monitoring heterogeneous catalytic reactions at electrochemical interfaces at high spatiotemporal resolution, several analytical approaches have been developed to

dynamically explore the heterogeneous interfacial structure–property relationship.<sup>13,14</sup> For instance, scanning probe techniques—including scanning electrochemical cell microscopy (SECCM),<sup>15</sup> scanning electrochemical microscopy (SECM),<sup>16</sup> and scanning photocurrent microscopy (SPCM)<sup>17</sup>—permit direct localized (photo)electrochemical measurements at high spatial resolution; however, they are somewhat ineffective at monitoring highly dynamic processes and providing a global overview of the processes. Nevertheless, optics-based electrochemical imaging methods—such as total internal reflection fluorescence microscopy (TIRFM),<sup>18–20</sup> electroluminescence microscopy (ECLM),<sup>21–24</sup> surface plasmon resonance microscopy (SPRM),<sup>25,26</sup> dark-field microscopy (DFM),<sup>27,28</sup> and interference reflection microscopy (IRM)<sup>29</sup>—can viably provide global information on dynamic processes at high spatiotemporal resolution owing to their widefield imaging mechanism. Recently, simple trans-illumination<sup>30</sup> and epi-illumination<sup>31</sup> imaging modes have been used to monitor electrochemical interfacial phenomena in *operando* by leveraging the absorption/reflection-related differences among various materials and metals.<sup>32</sup>

In particular, methods for characterizing heterogeneous photoelectrocatalytic properties are mainly rooted in two approaches: Scanning Photocurrent Microscopy (SPCM)<sup>33</sup> and molecular reaction imaging.<sup>34</sup> SPCM can determine spatially resolved photocurrents by scanning a diffraction-limited

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focused laser spot across the sample surface. However, owing to carrier diffusion and a deviation from the operating condition (widefield excitation), the obtained information might not entirely reflect the intrinsic heterogeneous structure–property relationship. In contrast, the widefield-imaging-based molecular reaction imaging method uses chromogenic or fluorogenic molecules to localize different types of active sites, and the imaging mechanism is compatible with both widefield excitation and point excitation. For instance, iodide oxidation reaction into the photocurrent microscope to map surface reactions induced by photogenerated holes in conjunction with bright-field imaging mode.<sup>35–37</sup>

Recently, a super-resolution single-molecule reaction labeling approach based on fluorogenic reaction of resorufin based molecules was established for nanometer-level determination of electrons (obtained *via* reduction of weakly fluorescent resazurin) or holes (generating *via* oxidation of nonfluorescent dihydroresorufin) for the wide-bandgap semiconductors TiO<sub>2</sub> and BiVO<sub>4</sub> under widefield laser illumination. This method was employed by Chen and co-workers to map the surface reactions induced by photogenerated holes or electrons.<sup>38,39</sup> Additionally, Fang and co-workers spatially resolved the heterogeneous dynamics of photocatalytic processes at individual defects in 2D layered InSe using single-molecule-labeling based on 3-aminophenyl fluorescein (APF), a very weakly fluorescent reactant molecule can be oxidized to a highly fluorescent product molecule fluorescein.<sup>40</sup> Notably, these labeling techniques can also be used to map the active sites in electrocatalytic processes.

However, despite the advantages of the super-resolution single-molecule-labeling method, the high-power laser excitation of single dye molecules (resorufin:  $E_{\text{max}} = 571 \text{ nm}$ , 2.17 eV; fluorescein:  $E_{\text{max}} = 491 \text{ nm}$ , 2.52 eV) might interfere with the operation of narrow-bandgap photoelectrocatalytic materials such as MoS<sub>2</sub> (bandgap =  $\sim 1.3 \text{ eV}$ ). Therefore, a facile widefield imaging method for mapping heterogeneous electron distribution in narrow-bandgap photoelectrocatalytic materials remains lacking. Moreover, given that TMD materials typically exhibit activity in both electrocatalysis and photoelectrocatalysis, studies have rarely been conducted to map the distribution of their active sites and elucidate the spatiotemporal relationship for the two processes.

In this study, a simple *in situ* widefield imaging method was developed to map heterogeneous electron transfer dynamics for different types of 2D semiconductors under electrocatalytic and photoelectrocatalytic conditions. Simple low-illumination-power imaging of an interfacial metal reduction reaction in trans and epi modes, which cause minimal disturbance to the photoelectrocatalytic process, aided in implementing the reporting mechanism. Cu(NH<sub>3</sub>)<sub>4</sub><sup>2+</sup> was employed as a reaction labelling molecule to map the electron distribution of electrons in both processes at high spatiotemporal resolution. The low-power widefield imaging strategy reported herein is fully compatible with the widefield excitation of photoelectrocatalytic processes, which are considerably more similar to the working conditions than those of laser-scanning methods.

The results indicated that the electron-transfer-associated active sites on n-type MoS<sub>2</sub> and p-type WSe<sub>2</sub> responded differently to electrocatalytic and photoelectrocatalytic conditions. For the n-type semiconductor, the electrocatalytic effect was significant but the photoelectrocatalytic effect was not obvious. For the p-type semiconductors, the photoelectrocatalytic effect shifted the onset potential forward by 680 mV and generated more active sites on the surface, promoting the catalytic reaction. Moreover, electron transfer in the WSe<sub>2</sub>-driven photoelectrocatalytic and electrocatalytic processes displayed a unique complementary pattern, suggesting intrinsic differences between the processes.

## Result and discussion

### (Photo)electrodeposition on n-type MoS<sub>2</sub>

Distributions of electrons and holes cannot be directly observed by regular optical microscopy. Therefore, the reduction of Cu(NH<sub>3</sub>)<sub>4</sub><sup>2+</sup> to metallic Cu, which can be detected optically in a straightforward manner owing to its strong scattering properties, was targeted in this study to probe the electron distribution (see ESI† for detailed information on the reduction reaction). The experimental setup constructed in this study (Fig. 1) was designed to accommodate a three-electrode electrochemical cell. The working electrode, comprising modified indium tin oxide (ITO) decorated with nanosheets of n-type MoS<sub>2</sub> or p-type WSe<sub>2</sub> (Fig. S1†), was mounted on the stage of an inverted fluorescence microscope, whereas Pt wire and a Hg/HgO electrode were used as the counter electrode and reference electrode, respectively. This setup was biased towards negative potentials, allowing *in situ* observation of the reduction behavior of Cu on the nanosheets. All potentials were measured against the Hg/HgO reference electrode and converted to those *versus* the reversible hydrogen electrode (RHE) as following<sup>41</sup> (Fig. S2†):

$$E_{\text{RHE}} = E_{\text{appl}} + 0.0592 \times \text{pH} + E_{\text{Hg/HgO}}^{\theta} \quad (1)$$

where  $E_{\text{appl}}$  is the measured potential *vs.* Hg/HgO,  $E_{\text{Hg/HgO}}^{\theta}$  is the standard potential of the reference electrode (0.098 V for

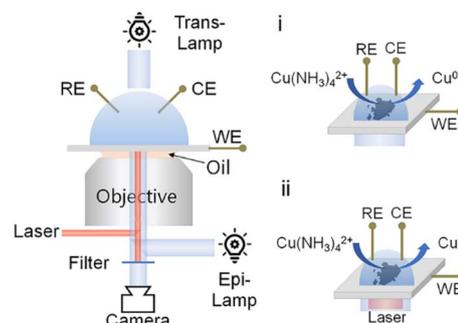


Fig. 1 Schematic of the electrochemical and optical system designed in this study for molecular reaction imaging. (i) Electrodeposition (ED) with electricity. (ii) Photoelectrodeposition (PED) using electricity and 640 nm widefield laser irradiation. WE, working electrode; RE, reference electrode; CE, counter electrode.



Hg/HgO in 1 M KOH), and pH = 9.7 for 1 M  $[\text{Cu}(\text{NH}_3)_4]\text{SO}_4 \cdot x\text{H}_2\text{O}$ .

Additionally, an optical system was designed for molecular reaction imaging in two different modes (Fig. 1): electrochemical deposition (ED) and photoelectrochemical deposition (PED). In the ED experiments, the processes were observed in epi mode. In the PED studies, considering the bandgaps of  $\text{MoS}_2$  and  $\text{WSe}_2$  ( $\sim 1.30$  eV and 1.25 eV, respectively), 640 nm laser irradiation was employed to excite electrons from the valence band to the conduction band and yield photogenerated charge carriers while the processes were being monitored in epi mode. Using a sensitive sCMOS camera, the power of the illumination out of the objective was maintained at a sufficiently low level ( $1 \mu\text{W}$ ;  $3.5 \text{ mW cm}^{-2}$ ) that it was insufficient to excite the material to produce photogenerated charge carriers in the experiments.

The ITO electrode coated with n-type  $\text{MoS}_2$  nanosheets, which were mechanically exfoliated from their crystal forms using low-residue Nitto tape, was investigated using bright-field transmission and reflection images (see Fig. 2a and c for representative images and Fig. S3† for additional images). The mechanical exfoliation resulted in nanosheets with different shapes and thicknesses (Fig. S4 and S5†). In the bright-field transmission images, regions with thicker and thinner nanosheets exhibited lower and higher transmission intensities,

respectively, because the thicker  $\text{MoS}_2$  nanosheets absorbed more light and exhibited lower transmittance.<sup>17</sup> Furthermore, the reflection intensities of nanosheets with different thicknesses varied in the reflection images. A combined analysis of the transmission and reflection images revealed that regions with high transmission intensity also exhibited high reflection intensity, that is, the thinner region of the nanosheets showed a higher reflection intensity (Fig. S6†).

The heterogeneity induced by mechanical exfoliation underpinned the operation of our single-nanosheet molecular reaction imaging method, which was developed to investigate the reactivity of “hot spots” for electron reduction on nanosheets *via* ED and PED. Cyclic voltammetry (CV) experiments were conducted to examine the Cu reduction behavior on the surface of the nanosheets *in situ*. To the end, the potential was swept from 0 to  $-1$  V at  $20 \text{ mV s}^{-1}$  (Fig. S7†). No significant differences were observed in the CV curve shape and current in the presence or absence of laser irradiation, presumably owing to the small number of nanosheets and the limited laser irradiation range. Also, because the nanosheets are mechanically exfoliated, the obtained samples have different thicknesses (Fig. S8a†). The reflection intensity of the nanometer-thick samples could vary significantly. The reflection of the thinner regions is normally higher (area 1 in Fig. S8c,†  $\sim 8300$ ), and that of the thicker regions is normally lower (area 2 in Fig. S8c,†

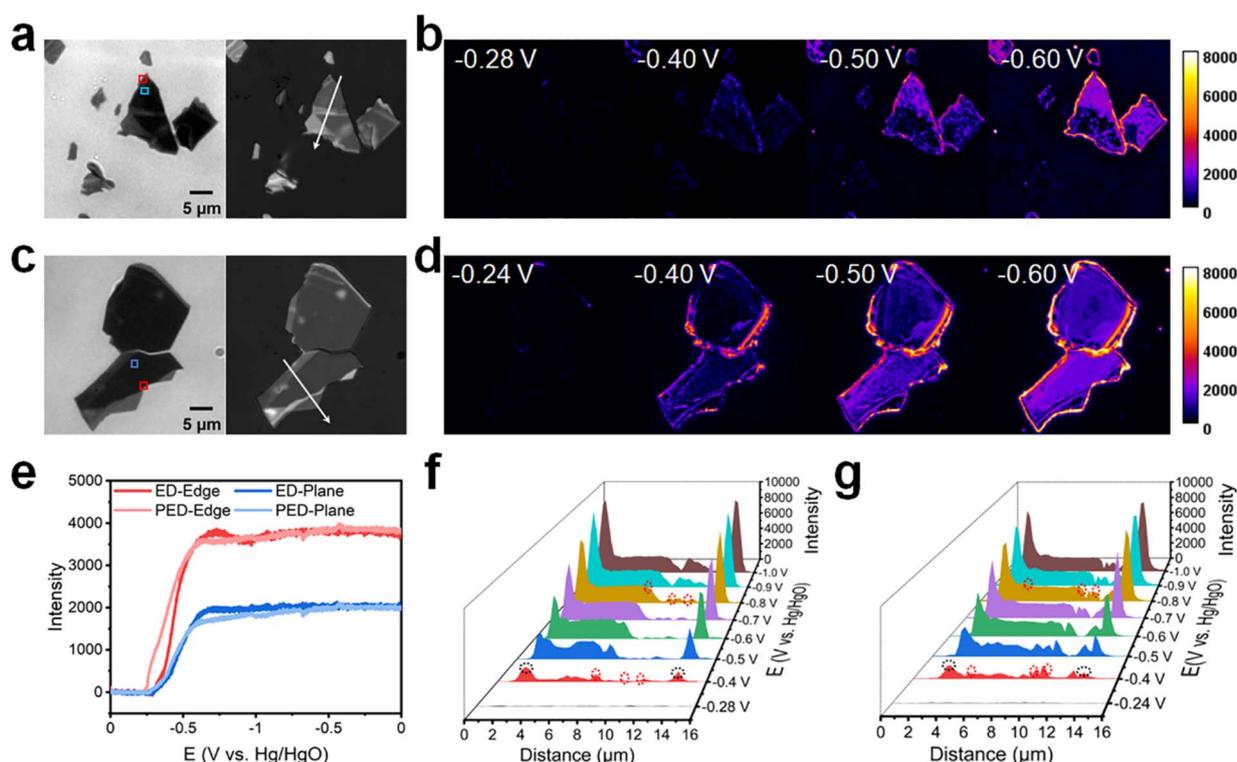


Fig. 2 Molecular reaction imaging of (photo)electrodeposition on n-type  $\text{MoS}_2$  nanosheets in 1 M  $[\text{Cu}(\text{NH}_3)_4]\text{SO}_4 \cdot x\text{H}_2\text{O}$  (pH = 9.7). (a and c) Bright-field transmission images (left) and reflection images (right) of the  $\text{MoS}_2$  nanosheets that underwent (a) ED and (c) PED. (b) ED of  $\text{MoS}_2$  nanosheets in epi mode. (d) PED of  $\text{MoS}_2$  nanosheets in epi mode and widefield 640 nm laser irradiation to stimulate  $\text{MoS}_2$  to generate carriers. (e) Reflection intensity–potential curves of the rectangular areas marked in (a) and (c), with the red and blue boxes representing an edge and plane, respectively. (f and g) Reflection intensity profiles acquired along the white arrows in (a) and (c), respectively, at different potential ( $E$ ) values. Black and red circles in (f) and (g) represent the positions of the perimeter edges and interior steps, respectively.



~5300). At the end of the reaction, we obtained the same reflection intensity (~7400) in the deposited Cu region (Fig. S8b†). Therefore, in the process of Cu deposition, the high-intensity regions became dimmer after Cu deposition, whereas the low-intensity region brightened (Fig. S8c†).

To clearly observe the Cu deposition on the surface of the nanosheets, a “difference” operation was performed to subtract background data. Molecular reaction imaging was conducted for ED on the ITO electrode coated with n-type MoS<sub>2</sub> nanosheets in a 1 M [Cu(NH<sub>3</sub>)<sub>4</sub>]SO<sub>4</sub>·xH<sub>2</sub>O electrolyte in epi mode (Fig. 2b). Notably, multilayer MoS<sub>2</sub> exhibits a narrow bandgap ( $E_g = 1.30$  eV) and thereby shows activity from the IR to UV regions.<sup>42</sup> Furthermore, the ITO electrode is optically transparent and does not absorb photons in the visible region. Therefore, 640 nm laser irradiation was employed in this study to activate the n-type MoS<sub>2</sub> to produce photogenerated carriers. Molecular reaction imaging was also conducted for PED on the ITO electrode coated with n-type MoS<sub>2</sub> nanosheets in a 1 M [Cu(NH<sub>3</sub>)<sub>4</sub>]SO<sub>4</sub>·xH<sub>2</sub>O electrolyte (Fig. 2d).

Intensity–potential curves were acquired for the different regions on the surface of MoS<sub>2</sub> during ED and PED (Fig. 2e; data extracted from Fig. 2b and d). During ED and PED, metallic Cu began to deposit at approximately  $-0.28$  V and  $-0.24$  V, respectively, with the difference in the onset potentials of Cu deposition being only 40 mV. Moreover, the intensity tended to stabilize at approximately  $-0.65$  V, indicating that Cu further extended outward after being sufficiently deposited. During deposition, the intensity changes at the edges were more significant than those on the planes.

Subsequently, the Cu deposition sites were investigated to further understand the manner in which surface heterogeneity affected Cu deposition. Reflection intensity profiles corresponding to ED were acquired (Fig. 2f) along the white arrow in Fig. 2a. Initially, no deposition occurred on the nanosheets at  $-0.28$  V. However, Cu began to nucleate as the voltage became more negative. At  $-0.4$  V, significant deposition was observed at the perimeter edges (1.9 and 14.1  $\mu\text{m}$ ; black dashed circles in Fig. S6a† and 2f), with minimal deposition occurring at the interior steps (7.7, 9.3, and 10.9  $\mu\text{m}$ ; red dashed circles in Fig. S6a† and 2f). On the plane, the Cu nucleation sites were concentrated in the thicker regions. As the voltage became more negative, Cu grew at the nucleation sites and extended outward.

Reflection intensity profiles corresponding to PED (Fig. 2g) were acquired along the white arrow in Fig. 2c. Initially, no deposition was observed on the nanosheets at  $-0.24$  V. However, Cu began to nucleate as the voltage became more negative. At  $-0.4$  V, significant deposition occurred at the perimeter edges (1.9 and 13.3  $\mu\text{m}$ ; black dashed circles in Fig. S6b† and 2g), with minimal deposition observed at the interior steps (4.5, 9.3, and 10.7  $\mu\text{m}$ ; red dashed circles in Fig. S6b† and 2g). Overall, the experimental results showed that illumination had no obvious effect on n-type MoS<sub>2</sub>.

In n-type semiconductors, the concentration of electrons is higher than that of holes; therefore, the Fermi level is near the conduction band.<sup>43,44</sup> Under illumination, the concentration of the photogenerated holes is significantly higher than the original equilibrium concentration; furthermore, the quasi-Fermi

level of holes deviates more significantly from the original Fermi level, whereas that of electrons is generally near the original Fermi level.<sup>45</sup> For the Cu deposition reaction, electrons are the charge carriers of interest. When a negative potential is applied, the holes produced under illumination pass through the external circuit to the counter electrode, whereas the electrons concentrate on the surface and participate in the surface reaction.<sup>46</sup> Because electrons are the majority carriers in n-type semiconductors, the electron concentration does not change significantly under illumination; therefore, the onset potential of Cu deposition does not show a significant positive shift. Additionally, when Cu is deposited on the surface of MoS<sub>2</sub>, it tends to nucleate at the perimeter edges and interior steps first and then nucleate at a more negative voltage on the thicker plane. This action mechanism is consistent with that reported previously.<sup>37,40,47</sup>

### (Photo)electrodeposition on p-type WSe<sub>2</sub>

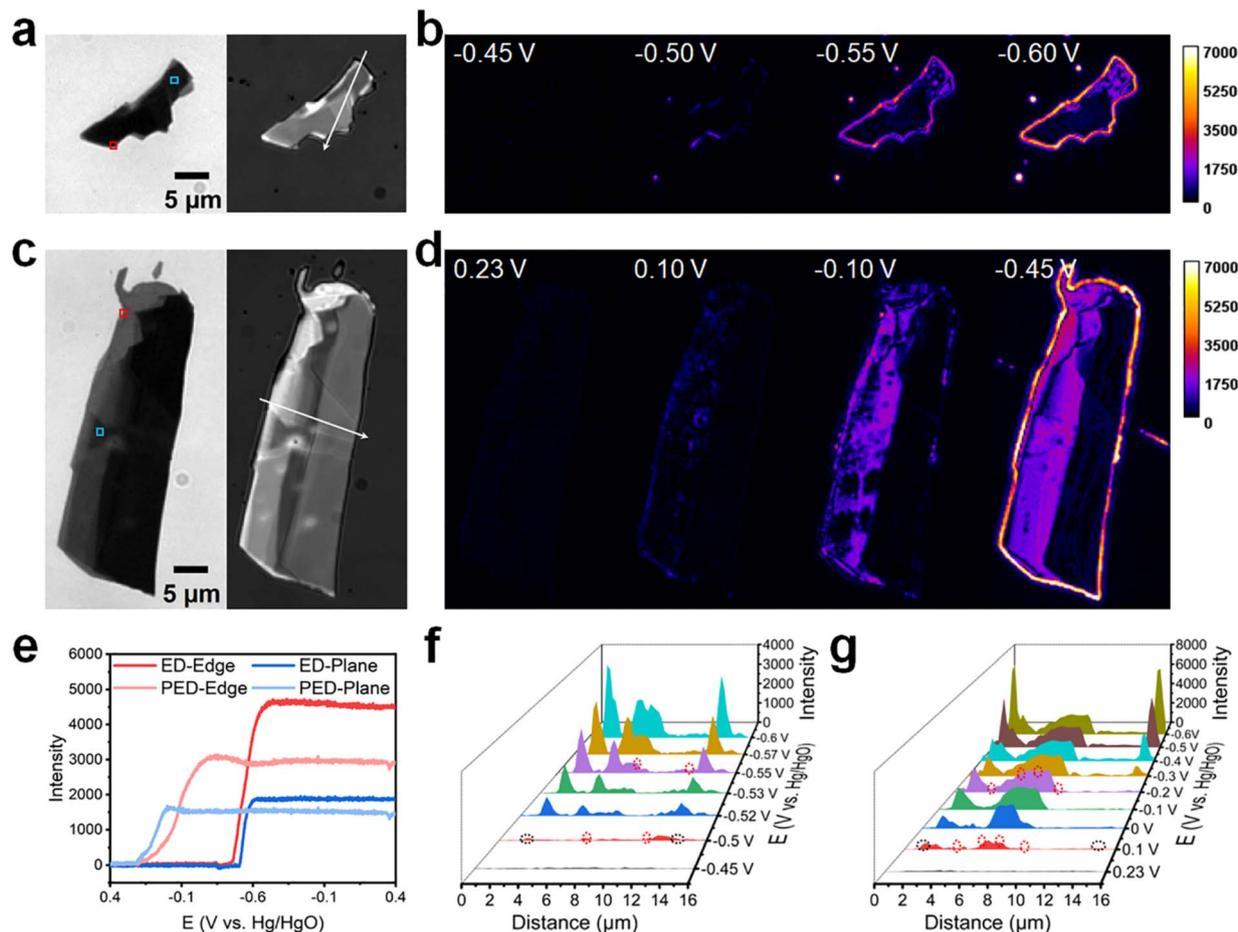
To gain deeper insights into the effects of surface heterogeneity on the Cu deposition, further studies were conducted using p-type WSe<sub>2</sub> nanosheets, which were prepared in the same manner as MoS<sub>2</sub> (mechanical exfoliation from crystals using low-residue Nitto tape; see Fig. S3† for additional images). The transmission and reflection intensities of p-type WSe<sub>2</sub> showed a correlation (Fig. S9†), indicating that regions with high transmission intensity also exhibited high reflection intensity. Moreover, the variation in the reflection intensity was more pronounced than that in the transmission intensity, enabling a better distinction between edges and planes.

Subsequently, molecular reaction imaging was conducted for ED and PED on the p-type WSe<sub>2</sub> nanosheets in epi mode (Fig. 3b and d). Owing to the narrow bandgap of multilayer WSe<sub>2</sub> ( $E_g = 1.25$  eV),<sup>48</sup> 640 nm laser irradiation was employed to activate p-type WSe<sub>2</sub> to produce photogenerated carriers. In the CV experiments, the potential was swept from 0.4 to  $-0.6$  V at 20 mV s<sup>-1</sup> (Fig. S10†). Furthermore, intensity–potential curves were acquired for different regions on the surface of WSe<sub>2</sub> during ED and PED (Fig. 3e; data extracted from Fig. 3b and d).

In ED, metallic Cu began to deposit at approximately  $-0.45$  V, a more negative value than that observed for n-type MoS<sub>2</sub>. Additionally, deposition at the edge occurred 50 mV earlier than that on the plane. The intensity changed rapidly but did not stabilize at the vertex potential of  $-0.6$  V; however, it settled at approximately  $-0.56$  V upon sweep reversal at  $-0.6$  V. In PED, metallic Cu began to deposit at  $\sim 0.23$  V; therefore, the difference in the onset potentials of Cu deposition was 680 mV. Moreover, the intensity at the plane and edge tended to stabilize within 250 mV and 550 mV, respectively. In the presence of laser irradiation, the growth rate of Cu on the plane exceeded that at the edge. Notably, the intensity change at the edge during deposition was more significant than that on the plane, similar to that of MoS<sub>2</sub>.

To directly observe Cu deposition on the sample surface, reflection intensity profiles were acquired along the white arrows shown in Fig. 3a and c for ED and PED (Fig. 3f and g). In ED, no deposition was initially observed on the nanosheets at





**Fig. 3** Molecular reaction imaging of (photo)electrodeposition on p-type WSe<sub>2</sub> nanosheets in 1 M [Cu(NH<sub>3</sub>)<sub>4</sub>]SO<sub>4</sub>·xH<sub>2</sub>O (pH = 9.7). (a and c) Bright-field transmission images (left) and reflection images (right) of the WSe<sub>2</sub> nanosheets that underwent (a) ED and (c) PED. (b) Electrodeposition of WSe<sub>2</sub> nanosheets in epi mode. (d) Photoelectrodeposition of WSe<sub>2</sub> nanosheets in epi mode and widefield 640 nm laser irradiation for stimulating WSe<sub>2</sub> to generate carriers. (e) Reflection intensity–potential curves of the rectangular areas marked in (a) and (c), with the red and blue boxes representing an edge and plane, respectively. (f and g) Reflection intensity profiles acquired along the white arrows in (a) and (c), respectively, at different *E* values; black and red circles represent positions of the perimeter edges and interior steps, respectively.

−0.45 V. However, Cu nucleation began as the voltage became more negative. At −0.5 V, moderate deposition occurred at the perimeter edges (1.6 and 13.3 μm; black dashed circles in Fig. S9a† and 3f) and interior steps (6.9 and 11.5 μm; red dashed circles in Fig. S9a† and 3f). Concurrently, Cu began to deposit on the plane, with nucleation sites concentrated in the thicker regions. In PED, significant deposition occurred at 0.1 V at the perimeter edges (1.3 and 15.2 μm; black dashed circles in Fig. S9b† and 3g) and interior steps (3.7, 5.9, 7.1 and 8.7 μm; red dashed circles in Fig. S9b† and 3g). The intensity changed significantly faster on the plane than at the edge. At 0 V, the intensity in the 4.8–8.8 μm region appeared to be higher than that in the 1–2 μm zone. Moreover, the intensity in the 4.8–8.8 μm region showed no significant change after −0.1 V, whereas that in the 1–2 μm zone increased until −0.6 V. Cu growth ceased once the deposition site on the plane was covered, whereas deposition continued on the deposited metallic Cu at the perimeter edge, enabling a relatively high-resolution assessment of electron distribution (~0.35 μm; Fig. S11†).

Furthermore, electron transfer on the WSe<sub>2</sub> surface was investigated using a focused laser spot at a wavelength of 640 nm, and the laser spot diameter was 0.64 μm. To the end, bright-field transmission and reflection images of WSe<sub>2</sub> were acquired (Fig. 4a; see Fig. 4a(iii) for the image of the focused laser spot); moreover, the Cu deposition on the nanosheets was analyzed when the focused laser spot was aimed at different sample locations (white crosses in Fig. 4b). In these experiments, a linear sweep voltammetric (LSV) potential was applied from 0.4 V to −0.1 V at 5 mV s<sup>−1</sup> and the focused laser density was set at 10 kW cm<sup>−2</sup>. Under these conditions, neither electricity nor laser irradiation enabled Cu deposition on the surface of WSe<sub>2</sub>. Nevertheless, Cu was preferentially deposited at an active site near the focused laser spot, indicating limited transport possibly because of electron capture by defects or electron consumption by recombination with holes during transport.<sup>9</sup>

However, a few electrons traveled to locations far from the laser-targeted areas (orange circles in Fig. 4b). Photogenerated



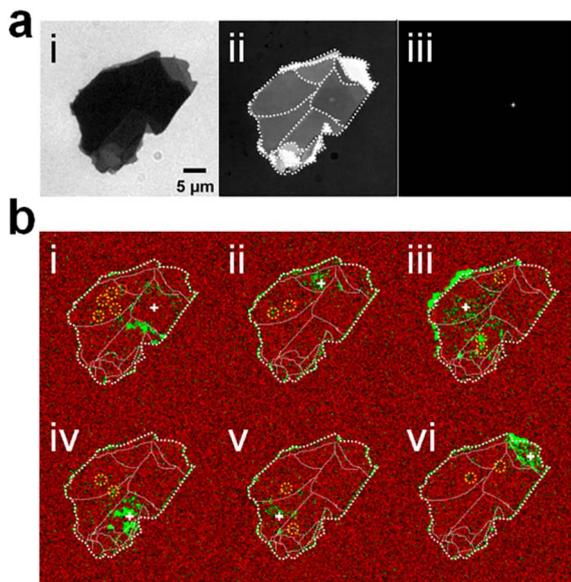


Fig. 4 Molecular reaction imaging of PED on p-type WSe<sub>2</sub> nanosheets with focused laser irradiation in 1 M [Cu(NH<sub>3</sub>)<sub>4</sub>]SO<sub>4</sub>·xH<sub>2</sub>O (pH = 9.7). (a) Bright field transmission (i) and reflection images (ii) of the WSe<sub>2</sub> nanosheets, with an image of the focused laser spot (iii). (b) PED of WSe<sub>2</sub> nanosheets in epi mode and 640 nm focused laser irradiation for stimulating WSe<sub>2</sub> to generate carriers. White dashed lines represent the nanosheet contour and interior steps. White crosses represent the laser-targeted locations. Orange circles represent certain deposition locations away from the laser location-targeted areas.

electrons moved tens of microns in a manner parallel to the layers from the carrier generating sites to the reactive hot spots; this distance is smaller than that reported by Samber and co-workers (45 μm).<sup>37</sup> Additionally, the onset potential of Cu deposition fluctuated with changes in the laser spot positioning. When the laser spot was aimed at a location (such as the white cross in Fig. 4b(iii)), Cu precipitation occurred at ~0.36 V; however, when it was directed at other sites (such as the white cross in Fig. 4b(v)), Cu started to deposit minimally at ~0.05 V, suggesting the uneven distribution of WSe<sub>2</sub> surface-active sites. Focused laser spots can induce high local catalytic activity owing to localized carrier injection;<sup>49</sup> however, this deviates from the actual working conditions and can therefore only guide studies on material modification. In contrast, wide-field laser irradiation is more suitable for evaluating the overall nanosheet activity *in operando*.

For p-type semiconductors, in which the hole concentration surpasses the electron concentration, the Fermi level is close to the valence band.<sup>43,44</sup> In the absence of light, electrons cannot readily transition to the conduction band, hindering further redox reactions on the surface. Consequently, the onset potential of p-type WSe<sub>2</sub> during ED is more negative than that of n-type MoS<sub>2</sub>. However, under illumination, the concentration of the photogenerated electrons significantly exceeds the original equilibrium concentration, causing a substantial deviation of the quasi-Fermi level of electrons from the original Fermi level. Furthermore, the quasi-Fermi level of holes typically remains near the original Fermi level.<sup>45</sup>

As mentioned earlier, electrons serve as charge carriers during Cu deposition reaction. When a negative potential is applied, the holes produced under illumination traverse the external circuit to the counter electrode, while electrons accumulate on the surface and participate in surface reactions.<sup>46</sup> Because electrons are the minority carriers in p-type semiconductors, their concentration varies significantly under illumination, leading to the noteworthy positive shift (680 mV) in the onset potential of Cu deposition observed in this study. The incident photons under laser action facilitate the formation of new active sites on the surface of WSe<sub>2</sub> owing to the incident photon, thereby generating additional electrons for Cu reduction, leading to a higher growth rate of Cu on the plane than that at the perimeter edge.<sup>50</sup>

Subsequently, the effects of laser density were investigated (Fig. 5) by varying the illumination density from 0 to 0.25 kW cm<sup>-2</sup>, and conducting molecular reaction imaging of the Cu reduction reaction. According to the results, the onset potential ( $E_{on}$ ) of Cu deposition on n-type MoS<sub>2</sub> (red curve in Fig. 5) increased slightly with increasing laser density, shifting from -0.30 V at 0 kW cm<sup>-2</sup> to -0.26 V at 0.25 kW cm<sup>-2</sup> (see Fig. S12† for detailed molecular reaction imaging data acquired at different laser densities). However, the  $E_{on}$  value of Cu deposited on p-type WSe<sub>2</sub> (black curve) shifted positively with increasing laser density, transitioning from -0.43 V at 0 kW cm<sup>-2</sup> to 0.23 V at 0.1 kW cm<sup>-2</sup>. Notably,  $E_{on}$  experienced only a slight positive shift when the laser power exceeded 0.1 kW cm<sup>-2</sup> (from 0.23 V at 0.1 kW cm<sup>-2</sup> to 0.27 V at 0.25 kW cm<sup>-2</sup>; see Fig. S13† for detailed molecular reaction imaging data obtained at different laser densities). The significant effect of the laser density stemmed from its ability to regulate the electron density.<sup>51</sup> As the light intensity increased, the carrier concentration saturated, resulting in no notable shift in  $E_{on}$  with a further increase in the light intensity.<sup>52</sup>

When a semiconductor is subjected to laser irradiation, the photogenerated carriers influence band bending at the

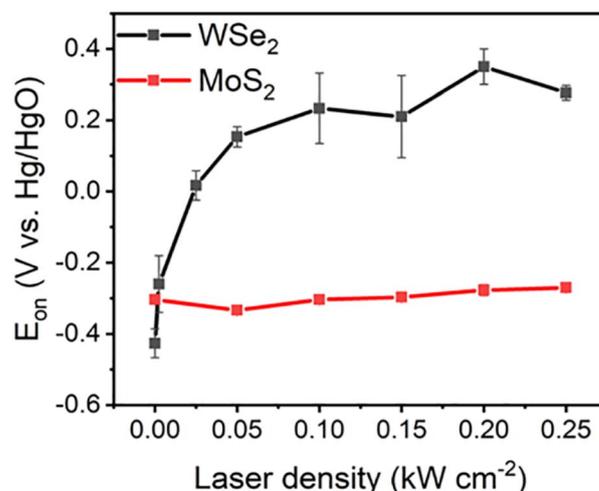


Fig. 5 Influence of laser density on onset potential ( $E_{on}$ ) of Cu deposition on two different nanosheets.



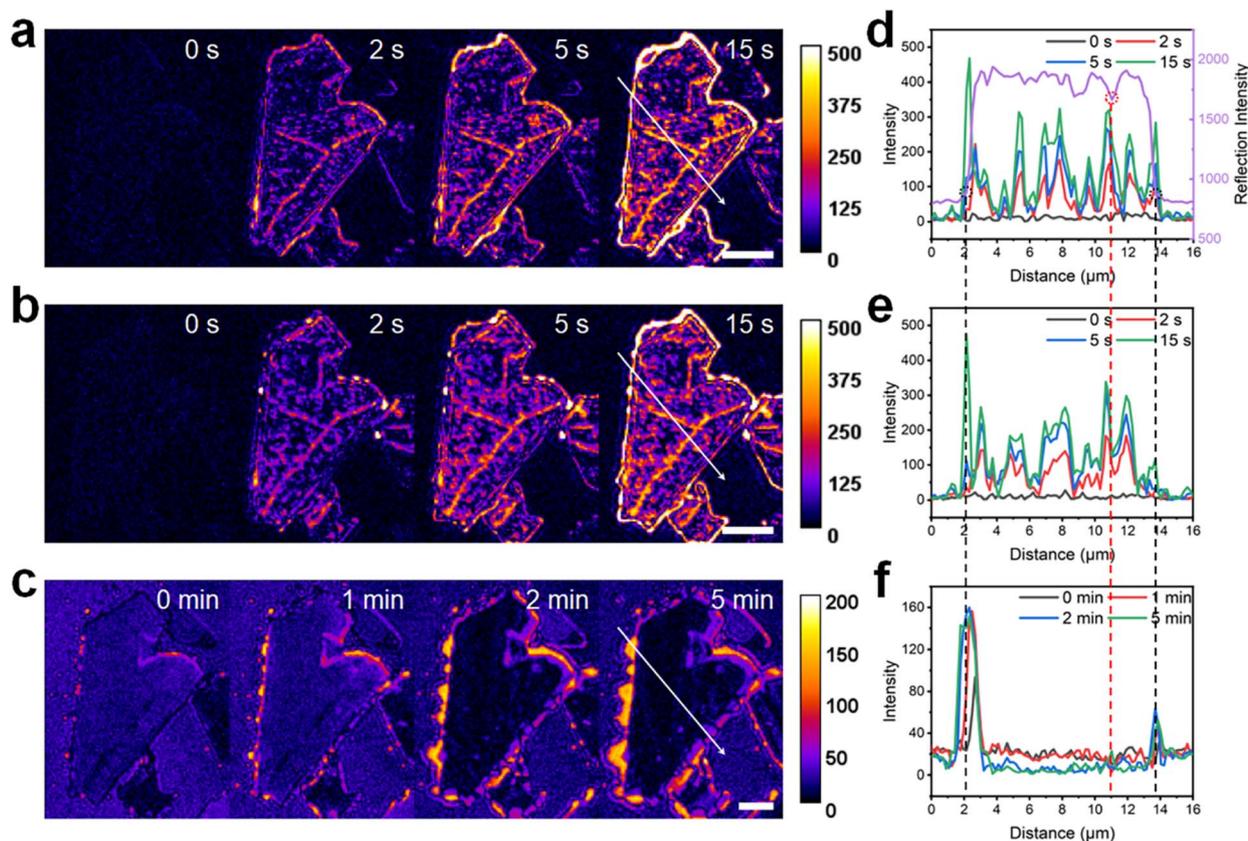


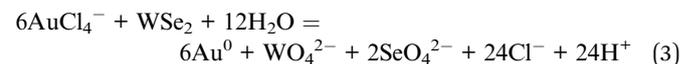
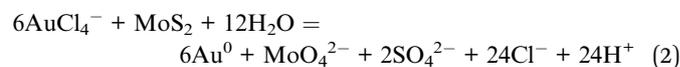
Fig. 6 Molecular reaction imaging of n-type MoS<sub>2</sub> nanosheets. (a) ED at a constant potential of  $-0.4$  V. (b) PED at a constant potential of  $-0.4$  V and laser density of  $0.25$  kW cm<sup>-2</sup>. (c) Chemical deposition in the absence of electricity and laser irradiation. (d–f) Reflection intensity profiles acquired at different durations along the white arrows shown in (a–c). Purple curve in (d) represents the reflection intensity of the primary reflection image shown in Fig. S16†. Black and red circles/dashed lines represent positions of the perimeter edges and interior steps, respectively. Scale bar:  $1$  μm.

semiconductor-solution interface, consequently affecting the open-circuit potential ( $V_{oc}$ ).<sup>53</sup> Therefore, the response of  $V_{oc}$  to light intensity was examined (Fig. S14†). Notably, the  $V_{oc}$  of n-type MoS<sub>2</sub> exhibited insignificant changes with respect to the light intensity, whereas that of p-type WSe<sub>2</sub> showed a positive shift. Additionally, the  $V_{oc}$  of the ITO substrate in the electrolyte remained unchanged as the light intensity was varied. Specifically, the  $V_{oc}$  of p-type WSe<sub>2</sub> underwent a positive shift from  $0.39$  V at  $0$  kW cm<sup>-2</sup> to  $0.54$  V at  $0.1$  kW cm<sup>-2</sup>. However, this shift was minimal when the laser density exceeded  $0.1$  kW cm<sup>-2</sup> (from  $0.54$  V at  $0.1$  kW cm<sup>-2</sup> to  $0.58$  V at  $0.25$  kW cm<sup>-2</sup>). Importantly, when the laser was deactivated,  $V_{oc}$  rapidly returned to  $\sim 0.38$  V. Conversely, for n-type MoS<sub>2</sub>,  $V_{oc}$  only experienced a  $10$  mV change after laser irradiation. These results agree with the changes observed in  $E_{on}$  with varying light intensity.

### Active site mapping via chemical deposition

To further investigate the influence of illumination on the semiconductor surface, HAuCl<sub>4</sub> was used as an oxidant to characterize the redox-active sites of the semiconductor. HAuCl<sub>4</sub> has been extensively used as a molecular probe to map chemically active sites.<sup>54–56</sup> The Fermi levels of MoS<sub>2</sub> and WSe<sub>2</sub>

are reportedly  $4.6$ – $4.9$  eV below the vacuum level.<sup>57</sup> Moreover, at  $0$  pH and  $1$  atm hydrogen pressure,  $E_{RHE}$  has been estimated to be  $4.43$ – $4.47$  eV below the vacuum level.<sup>58</sup> Additionally, the reduction potential of AuCl<sub>4</sub><sup>-</sup> ( $AuCl_4^- + 3e^- = Au^0 + 4Cl^-$ ) is  $+1.002$  V vs. the standard hydrogen electrode (SHE) potential. Thus, electrons in the nanosheets may spontaneously shift to HAuCl<sub>4</sub> to reduce it to metallic Au<sup>0</sup> particles while oxidizing themselves to soluble species in the aqueous solution, as expressed below.<sup>59</sup>



In the present study, the same nanosheet was used for ED, PED and chemical deposition. To eliminate the influence of light on the surface of the nanosheets, ED was first performed, followed by the complete dissolution of the deposited metallic Cu in accordance with reaction (S4),† which was evidently achieved within  $\sim 5$  min (Fig. S15†). Subsequently, PED was performed on the same nanosheet and the deposited metallic Cu was dissolved using the method described above. Finally,



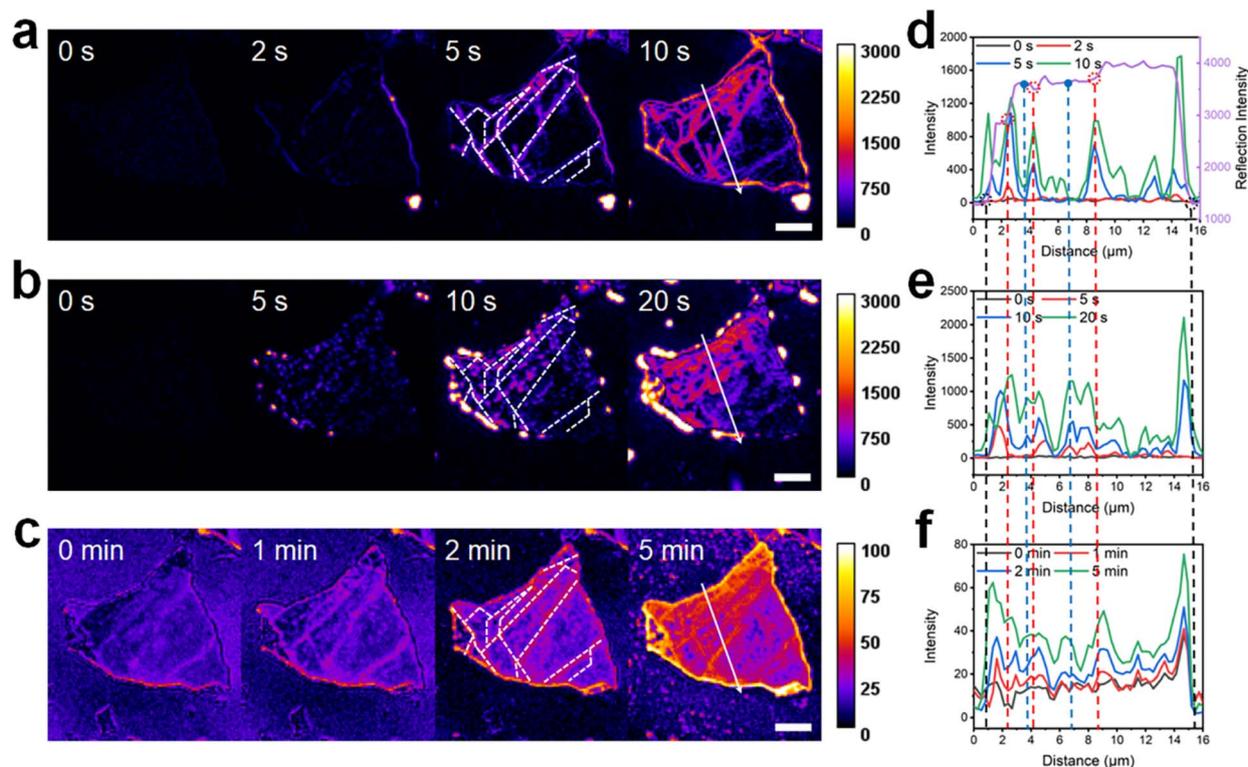


Fig. 7 Molecular reaction imaging of p-type  $\text{WSe}_2$  nanosheets. (a) ED at a constant potential of  $-0.6$  V. (b) PED at a constant potential of  $-0.1$  V and laser density of  $0.25 \text{ kW cm}^{-2}$ . (c) Chemical deposition in the absence of electricity and laser irradiation. White dashed lines represent interior steps. (d–f) Reflection intensity profiles acquired at different durations along the white arrows shown in (a–c). Purple curve in (d) denotes the reflection intensity of the primary reflection image shown in Fig. S16†. Black and red circles/dashed lines represent positions of the perimeter edges and interior steps, respectively. Blue spots/dashed lines denote positions of the plane. Scale bar:  $1 \mu\text{m}$ .

the redox sites on the nanosheets were characterized by chemical deposition. In this study, chemical reduction of  $\text{HAuCl}_4$  was monitored in trans-illumination mode to obtain a better contrast. The “difference” operation was conducted to subtract background data, thus permitting relatively unimpeded observation of the deposition sites.

Molecular reaction imaging was performed for ED, PED, and chemical deposition on the same n-type  $\text{MoS}_2$  nanosheet (Fig. 6), and the corresponding bright-field transmission and reflection images of the  $\text{MoS}_2$  nanosheets were acquired (Fig. S16a†). The results revealed that no significant differences in the distribution of the Cu deposition sites under dark and laser illumination conditions. The nucleation sites were primarily concentrated at the perimeter edges and interior steps and evenly distributed on the plane (Fig. 6d and e). As time progressed, the deposited Cu nanoparticles grew at the nucleation sites, and the coverage area of Cu on the semiconductor surface increased. However, in the chemical deposition experiments (Fig. 6c and f), Au particles were deposited only at the perimeter edge of the nanosheet, indicating that the redox sites on the surface of  $\text{MoS}_2$  existed only at the perimeter edge. Cu deposition on the surface during ED and PED was likely influenced by the electrocatalytic activation of certain low-activity sites on the surface, whereas the laser irradiation did not have a significant catalytic effect on n-type  $\text{MoS}_2$ .

Molecular reaction imaging was also performed for ED, PED, and chemical deposition on the same p-type  $\text{WSe}_2$  nanosheet (Fig. 7), and the corresponding bright-field transmission and reflection images were acquired (Fig. S16b†). The Cu electro-deposited on  $\text{WSe}_2$  at a constant potential of  $-0.6$  V mainly concentrated at the perimeter edges and interior steps in a linear manner (dashed white lines in Fig. 7a). Because the deposition potential of Cu shifted significantly in the presence of laser irradiation, PED was conducted at  $-0.1$  V (Fig. 7b). The results indicated that the nucleation sites of Cu were mainly located in the plane, with a few nucleation sites present at the perimeter edge, thereby exhibiting a point-like distribution. After dissolving metallic Cu, chemical deposition was performed on the same nanosheet (Fig. 7c). Au particles were deposited not only at the perimeter edge and the interior step of the nanosheet but also on the plane of the nanosheet, suggesting that the active sites of the electrocatalytic and photo-electrocatalytic redox reactions were combined.

The obtained reflection intensity profiles suggested that the peak intensities corresponding to the ED and PED were always complementary (Fig. 7d and e). Additionally, chemical deposition presumably occurred at the active sites of ED and PED (Fig. 7f). For example, at the interior steps ( $2.4$ ,  $4.2$ , and  $8.7 \mu\text{m}$ ; red dashed lines in Fig. 7d–f), the highest and lowest intensities were observed during ED and PED, respectively.



Conversely, at the plane (3.7, 6.8  $\mu\text{m}$ ; blue dashed lines in Fig. 7d–f), an intensity change was observed during PED but not during ED. Furthermore, at the aforementioned locations, significant intensity changes occurred during chemical deposition. This was probably due to the laser irradiation-induced generation of more new active sites on the surface of  $\text{WSe}_2$ , resulting in a higher more charge for Cu reduction. Consequently, the distribution of Cu nanoparticles on the surface of the semiconductors differed from each other.<sup>50</sup>

## Conclusion

An optical imaging strategy was developed to visualize the dynamic distribution of electron-transferring active sites on narrow-bandgap semiconductors *in situ* in different catalytic processes using metal deposition as the reporting mechanism. In this approach, n-type  $\text{MoS}_2$  and p-type  $\text{WSe}_2$  showed different responses in electrocatalytic and photoelectrocatalytic processes. For n-type  $\text{MoS}_2$ , a high electron concentration diminished the difference between scenarios with and without illumination. For p-type  $\text{WSe}_2$ , in which electrons were the minority carriers, the limited activation sites concentrated at the steps and edges in the electrocatalytic process. In the presence of light, the photogenerated electrons significantly shifted the reaction onset potential in a positive direction, producing more active sites with a uniform distribution on the  $\text{WSe}_2$  plane. Thus, an interesting complementary pattern was discovered for different processes. This simple, versatile analytical approach opens a new path for comprehensively clarifying the heterogeneous structure–property relationships of various electrocatalytic and photoelectrocatalytic materials, which could further aid the rational design of high-performance catalytic devices.

## Data availability

The experimental procedures and data are provided in the ESI.†

## Author contributions

R. H. conceived and supervised the project. L. W. and H. C. performed the experiments. L. W. and R. H. wrote the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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