

Cite this: *Chem. Sci.*, 2024, 15, 11902

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# Unraveling the atomic structure and dissociation of interfacial water on anatase TiO<sub>2</sub> (101) under ambient conditions with solid-state NMR spectroscopy†

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Anatase TiO<sub>2</sub> is a widely used component in photo- and electro-catalysts for water splitting, and the (101) facet of anatase TiO<sub>2</sub> is the most commonly exposed surface. A detailed understanding of the behavior of H<sub>2</sub>O on this surface could provide fundamental insights into the catalytic mechanism. This, however, is challenging due to the complexity of the interfacial environments, the high mobility of interfacial H<sub>2</sub>O, and the interference from outer-layer H<sub>2</sub>O. Herein, we investigate the H<sub>2</sub>O/TiO<sub>2</sub> interface using advanced solid-state NMR techniques. The atomic-level structures of surface O sites, OH groups, and adsorbed H<sub>2</sub>O have been revealed and the detailed interactions among them are identified on the (101) facet of anatase TiO<sub>2</sub>. By following the quantitative evolution of surface O and OH sites along with H<sub>2</sub>O loading, it is found that more than 40% of the adsorbed water spontaneously dissociated under ambient conditions on the TiO<sub>2</sub> surface at a loading of 0.3 mmol H<sub>2</sub>O/g, due to the delicate interplay between water–surface and water–water interactions. Our study highlights the importance of understanding the atomic-level structures of H<sub>2</sub>O on the surface of TiO<sub>2</sub> in catalytic reactions. Such knowledge can promote the design of more efficient catalytic systems for renewable energy production involving activation of water molecules.

Received 26th April 2024  
Accepted 25th June 2024

DOI: 10.1039/d4sc02768j

rsc.li/chemical-science

## Introduction

The behavior of water on the surface of metal oxide or metal is a subject of immense importance in a variety of fields such as heterogeneous catalysis, energy science, and materials science.<sup>1–8</sup> This is because water plays a critical role in various chemical processes that occur on the surface of these materials.<sup>9–19</sup> A detailed understanding of the structure and behavior of interfacial H<sub>2</sub>O on these surfaces can help

researchers to develop more efficient and effective catalytic systems and materials for energy storage and conversion.

One material that has been extensively studied in this regard is titanium dioxide (TiO<sub>2</sub>). TiO<sub>2</sub> is widely used in various fields, including photocatalysis, solar cells, and sensors.<sup>1–3,20–22</sup> The surface of TiO<sub>2</sub> is highly hydrophilic, and therefore water molecules readily adsorb onto the surface. However, details of the interaction of water with the surface, the reactivity of the TiO<sub>2</sub> surface, the rupture of the water O–H bond and consequently the formation of the water–oxide interface structure are essential for the understanding of the interfacial water behavior on the TiO<sub>2</sub> surface.

A number of studies have been conducted to investigate the behavior of water on the TiO<sub>2</sub> surface. By using scanning tunneling microscopy (STM), it was reported that water molecules form a highly ordered structure on the TiO<sub>2</sub> surface, and depending on the conditions, one/two/three-dimensional water structures (monomer, dimer, trimer, tetramer, chains, and networks) could be formed on the surface.<sup>23–27</sup> In other reports, it was suggested that a water molecule could react with the oxygen vacancy/defect or rupture over the low coordination surface Ti sites of TiO<sub>2</sub>, and forms two hydroxyls.<sup>11,28–31</sup> Very recently, by an elegant environmental TEM method, Wang *et al.* reported that the four-coordinated Ti on a (1 × 4) reconstructed

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† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4sc02768j>

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TiO<sub>2</sub> (001) surface is highly active for water activation.<sup>32</sup> However, water adsorption and dissociation on non-defect TiO<sub>2</sub> has been disputed for decades.<sup>33–36</sup> For the interfacial H<sub>2</sub>O on the (101) facet of anatase TiO<sub>2</sub>, a number of theoretical calculations<sup>37–40</sup> and some experimental studies, including temperature-programmed desorption (TPD)<sup>41</sup> and STM,<sup>23</sup> have concluded an intact molecular adsorption of H<sub>2</sub>O on the TiO<sub>2</sub> surface. In contrast, a partial dissociation of H<sub>2</sub>O has been inferred through the observation of hydroxyls formed on the TiO<sub>2</sub> surface by using spectroscopic techniques, such as X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), sum frequency generation (SFG) and so on.<sup>42–49</sup> However, it is challenging to distinguish the OH groups formed by H<sub>2</sub>O dissociation on the non-defect TiO<sub>2</sub> surface from either the OH groups generated by H<sub>2</sub>O reaction with defect sites or the original OH groups present on TiO<sub>2</sub>. Therefore, while progress has been made, the atomic-level structures of interfacial H<sub>2</sub>O and especially its detailed interaction with the TiO<sub>2</sub> surface are still to be resolved. Significantly, several challenges need to be addressed<sup>50</sup> to get the atomic-level structures of the interplay, including how to reveal the interaction of surface oxygen and titanium atoms with interfacial H<sub>2</sub>O, how the water splitting and hydroxyl formation occur, and the quantitative evolution of these surface oxygen species at the water/TiO<sub>2</sub> interface during the hydration process. All these problems are difficult to solve under working reaction conditions due to the interference from outer-layer H<sub>2</sub>O, the high mobility of interfacial H<sub>2</sub>O, and the complexity of interfacial environments.<sup>15,51,52</sup>

<sup>17</sup>O NMR has long been a powerful and attractive approach for characterizing the atomic-level structure of various oxygen-containing materials due to the wide <sup>17</sup>O chemical shift range.<sup>53–63</sup> For nano-oxides, the relatively low gyromagnetic ratio, low <sup>17</sup>O abundance (0.037%), quadrupolar nature, and especially low proportion of surface oxygen atoms lead to a great challenge to investigate the detailed structure of surface oxygen atoms. The sensitivity and resolution were still limited despite <sup>17</sup>O isotopic enrichment. Recently, Peng and Grey *et al.* selectively enriched surface oxygen atoms on metal oxides (including CeO<sub>2</sub>, TiO<sub>2</sub>, and ZnO) with H<sub>2</sub><sup>17</sup>O, and distinguished surface O/OH sites from bulk oxygen by using one-dimensional (1D) <sup>17</sup>O MAS NMR spectra combined with DFT calculations.<sup>64–66</sup> We identified surface OH groups and (sub-)surface O sites on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> by using the 2D proton-detected <sup>1</sup>H–<sup>17</sup>O heteronuclear correlation technique to improve the sensitivity and resolution of <sup>17</sup>O NMR spectra,<sup>67</sup> which would make it possible to study interactions between interfacial H<sub>2</sub>O and oxide surfaces. To date, the detailed structures of the surface O/OH sites and their interactions with interfacial H<sub>2</sub>O are still ambiguous, let alone quantitative evolution of the structure of these surface O/OH sites in the presence of interfacial H<sub>2</sub>O, which should be the key to understand the H<sub>2</sub>O–oxide interaction.

In this study, we choose anatase TiO<sub>2</sub> nanoparticles as the oxide model owing to their wide practical application in photo- and photoelectro-catalysis, on which the most frequently exposed surface is the (101) facet with the lowest energy, and investigated the ability of anatase (101) facet-dominated TiO<sub>2</sub> nanoparticles to adsorb and activate water on their surface. By

using the 2D <sup>17</sup>O MQMAS and <sup>1</sup>H{<sup>17</sup>O} J-HMQC NMR methods and other techniques, we examined the atomic-level structures of interfacial water and surface O/OH sites of TiO<sub>2</sub> at different water loading levels, as well as the through-bond interactions between them. Our findings show that due to the delicate interplay between water–surface and water–water interactions, the O–H bond of the adsorbed water is broken through the joint effort of coordination-unsaturated Ti<sub>5C</sub> and the adjacent surface O<sub>2C</sub> sites, resulting in a terminal OH group (Ti<sub>5C</sub>–OH) and a proton accommodated on the surface O<sub>2C</sub> site to form a bridging hydroxyl (O<sub>2C</sub>H). By following the quantitative evolution of surface O and OH sites along with H<sub>2</sub>O loading, it is demonstrated that at a loading of 0.3 mmol H<sub>2</sub>O g<sup>–1</sup>, over 40% of the adsorbed water was dissociated spontaneously on the TiO<sub>2</sub> surface. The understanding over the structure and behavior of interfacial H<sub>2</sub>O is helpful for developing more efficient and effective catalytic systems for energy storage and conversion.

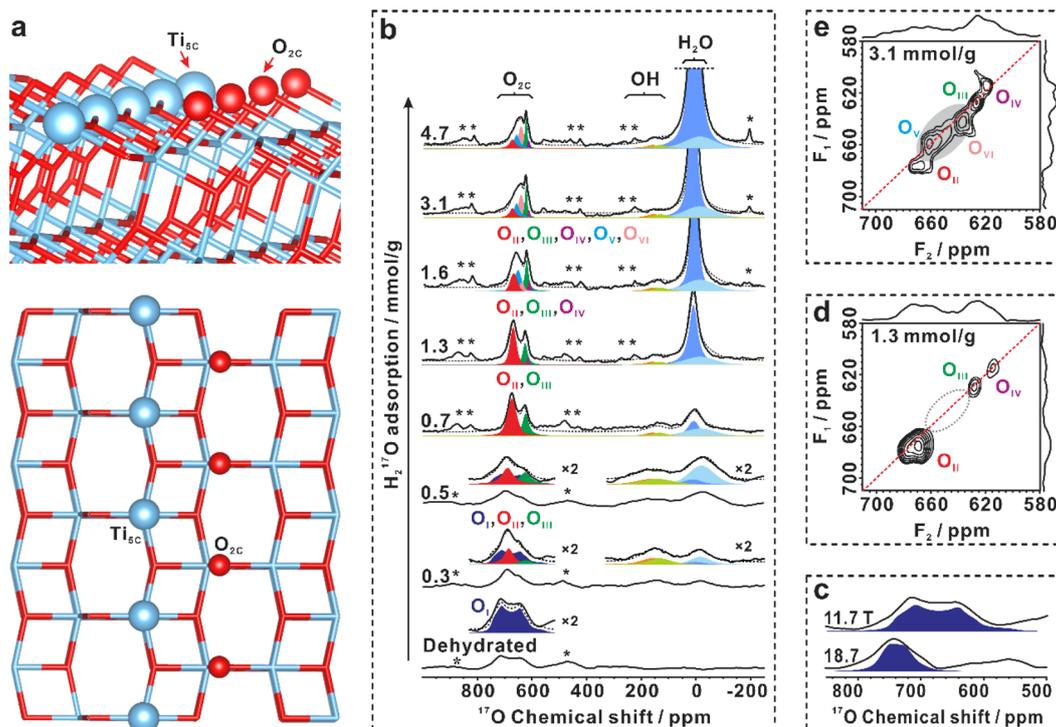
## Results and discussion

### Interaction of surface oxygen sites with H<sub>2</sub>O on the (101) facet of TiO<sub>2</sub>

Anatase TiO<sub>2</sub> (Fig. S1–S4 in ESI<sup>†</sup>) with predominantly (101) facets (95% percent) has been synthesized.<sup>68,69</sup> No oxygen vacancies were detected at liquid nitrogen temperature by the ESR spectra (Fig. S5<sup>†</sup>) before and after dehydration of the TiO<sub>2</sub> sample at 160 °C. The morphology of the (101) facet of anatase is well-documented.<sup>23</sup> It features a saw-tooth-like shape, and its top layer is composed of two coordination-unsaturated species in close proximity: the five-coordinated Ti (Ti<sub>5C</sub>) and two-coordinated surface O (O<sub>2C</sub>) atoms, as illustrated in Fig. 1a.

To investigate the surface sites on the (101) facet of TiO<sub>2</sub> that interact with H<sub>2</sub>O, one-dimensional (1D) and two-dimensional (2D) <sup>17</sup>O MAS NMR experiments were performed on <sup>17</sup>O-enriched TiO<sub>2</sub> (with the surface layer of TiO<sub>2</sub> enriched with <sup>17</sup>O, see methods for details) with different H<sub>2</sub><sup>17</sup>O loadings. As shown in Fig. 1b, the resonances at 600–800 ppm in the <sup>17</sup>O MAS NMR spectrum of bare TiO<sub>2</sub> were observed, the chemical shift of which is well above the three-coordinated oxygen in the bulk (around 400–600 ppm, Fig. S6<sup>†</sup>).<sup>65</sup> The signal shows a broad second-order quadrupolar interaction lineshape (11.7 T), with its line width sharply decreasing with the increase of magnetic field (18.7 T, Fig. 1c). We assign the resonance to a surface O<sub>2C</sub> site without interaction with H<sub>2</sub>O on bare TiO<sub>2</sub>, marked as O<sub>I</sub>. Upon loading 0.03 mmol H<sub>2</sub><sup>17</sup>O over 100 mg TiO<sub>2</sub> (0.3 mmol g<sup>–1</sup>), the shape of the signal at 600–800 ppm changes significantly, with a tip appearing at 700 ppm (Fig. 1b). At the same time, two new resonances centered at 0 and 150 ppm emerge. To gain insight into the structure of the surface oxygen sites interacting with H<sub>2</sub>O, we used the 2D <sup>17</sup>O 3Q MAS NMR technique to remove the quadrupolar broadening and enhance the spectral resolution. For TiO<sub>2</sub> loaded with 0.3 mmol g<sup>–1</sup> H<sub>2</sub><sup>17</sup>O, clearly, the 2D 3Q MAS spectrum resolved the overlapped resonances in the <sup>17</sup>O MAS NMR spectrum (Fig. S7<sup>†</sup>), revealing two new two-coordinated O<sub>2C</sub> sites (O<sub>II</sub> and O<sub>III</sub>). However, the large quadrupolar interaction of the O<sub>I</sub> site made it hardly





**Fig. 1** Surface  $O_{2C}$  sites interacting with  $H_2O$ . (a) Ball-and-stick model of the anatase  $TiO_2$  (101) surface with five-coordinated Ti ( $Ti_{5C}$ ) and two-coordinated surface O ( $O_{2C}$ ) atoms. Titanium and oxygen atoms are plotted in blue (Ti) and red (O). (b) 1D  $^{17}O$  MAS NMR spectra of dehydrated  $^{17}O$ -enriched  $TiO_2$  samples with  $H_2^{17}O$  loading from 0 to  $4.7 \text{ mmol g}^{-1}$ , acquired at a magnetic field of 11.7 T. Note that six  $O_{2C}$  sites ( $O_I$ – $O_{VI}$ ), two types of OH groups, and two types of adsorbed  $H_2O$  correspond to different  $^{17}O$  peaks marked by different colors. (c) 1D  $^{17}O$  MAS NMR spectra of  $^{17}O$ -enriched bare  $TiO_2$ , acquired at magnetic fields of 18.7 and 11.7 T. (d–e) 2D  $^{17}O$  3Q MAS NMR spectra of dehydrated  $^{17}O$ -enriched  $TiO_2$  samples with 1.3 (d) and  $3.1 \text{ mmol g}^{-1}$  (e)  $H_2^{17}O$  loading, acquired at a magnetic field of 11.7 T. Asterisks denote spinning sidebands.

observable in the 2D 3Q MAS spectrum. When the  $H_2^{17}O$  loading increased to  $1.3 \text{ mmol g}^{-1}$  and then to  $3.1 \text{ mmol g}^{-1}$ , up to five signals were identified, representing five types of  $O_{2C}$  sites ( $O_{II}$ ,  $O_{III}$ ,  $O_{IV}$ ,  $O_V$ , and  $O_{VI}$ ) (Fig. 1d and e). The appearance of new  $O_{2C}$  sites at the expense of  $O_I$  sites demonstrates that a fraction of two-coordinated surface oxygen sites can interact with water molecules, leading to changes in their coordination environments.

The NMR parameters of these  $O_{2C}$  sites ( $O_I$ – $O_{VI}$ ) obtained from 1D and 2D  $^{17}O$  3Q MAS NMR spectra (Table S1†) were used to deconvolute the 1D  $^{17}O$  MAS NMR signals acquired at 11.7 T and 18.7 T (Fig. 1b and S8†). The different oxygen species ( $O_I$ – $O_{VI}$ ) obtained upon water loading indicate different interplays between the  $O_{2C}$  sites and water. Notably, the  $^{17}O$  MAS NMR spectrum of bare  $TiO_2$  does not exhibit any signals of adsorbed  $H_2O$  or surface OH groups (usually at  $-200$  to  $200$  ppm) (Fig. 1b). At a water loading of  $0.3 \text{ mmol g}^{-1}$ , besides the change in  $O_{2C}$  sites, the adsorbed  $H_2O$  at  $-50$ – $10$  ppm and two new overlapped species of surface hydroxyls at around  $150$  ppm emerge (Fig. 1b), which can be well resolved by the following 2D  $^1H\{^{17}O\}$  J-HMQC NMR experiments. The appearance of hydroxyls suggests that water splitting occurs, which is also confirmed by  $^2H$  MAS NMR (see the following). These results demonstrate that: (1) water can interact with the surface  $O_{2C}$  site, resulting in a change in its chemical environment, although the type and strength of the interaction cannot be

determined at the present time; and (2),  $^{17}O$  MAS NMR experiments confirm the formation of hydroxyls arising from the dissociation of water on the  $TiO_2$  surface.

### Detailed interaction of surface oxygen/hydroxyl sites with interfacial $H_2O$

To better understand the structure of the interfacial scenario during water adsorption, a series of 2D  $^1H\{^{17}O\}$  J-HMQC NMR experiments were conducted. These experiments allowed for the identification of  $^1H$ – $^{17}O$  correlation/connectivity through chemical bonds and strong hydrogen bonds. There are three types of  $^1H$  NMR signals at 1.7, 5.4, and 7.0 ppm, which can be assigned to the proton of terminal hydroxyl, adsorbed  $H_2O$ , and bridging hydroxyl, respectively.<sup>13,14</sup> At  $H_2^{17}O$  loadings of 0.5, 1.3 and  $3.1 \text{ mmol g}^{-1}$ , the result showed that  $O_{2C}$  sites were strongly correlated with adsorbed  $H_2O$  (5.4 ppm, Fig. S9†), indicating that the surface coordination-unsaturated  $O_{2C}$  sites were responsible for the interaction with water. Interestingly, no correlation was observed between the  $O_{2C}$  sites and terminal/bridging hydroxyls, indicating that the hydrogen in surface hydroxyls is not connected to the adjacent  $O_{2C}$  site *via* hydrogen-bonds.

The spatial relationship between the adsorbed water and the two surface hydroxyls was also investigated. Fig. 2a and b display the  $^1H$ – $^{17}O$  correlations between adsorbed  $H_2O$  and



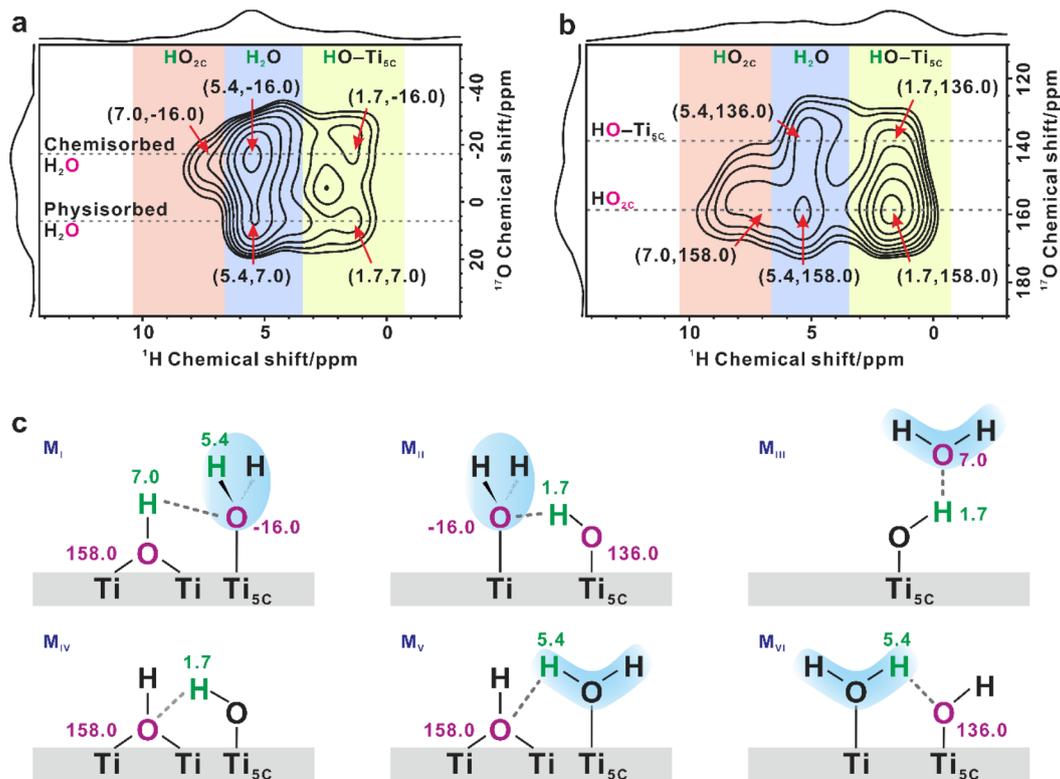


Fig. 2 Detailed structure of interfacial  $\text{H}_2\text{O}$  and OH groups on  $\text{TiO}_2$ . (a and b) 2D  $^1\text{H}$ ( $^{17}\text{O}$ ) J-HMQC NMR spectra of the dehydrated  $^{17}\text{O}$ -enriched  $\text{TiO}_2$  sample with  $0.5 \text{ mmol g}^{-1} \text{H}_2^{17}\text{O}$  loading in different  $^{17}\text{O}$  chemical shift ranges. (c) The schematic illustration of the different structures ( $M_{\text{I}}$ – $M_{\text{VI}}$ ) of surface hydroxyls and adsorbed  $\text{H}_2\text{O}$  molecules over the surface of  $\text{TiO}_2$ , with the  $^1\text{H}$  and  $^{17}\text{O}$  chemical shift values marked. Water molecule is shadowed in blue.

surface OH groups on  $\text{TiO}_2$  with a  $0.5 \text{ mmol g}^{-1} \text{H}_2^{17}\text{O}$  loading in different  $^{17}\text{O}$  chemical shift ranges. As shown in Fig. 2a, two cross peaks were visible at (5.4, -16) ppm and (5.4, 7.0) ppm in the correlation experiment, indicating the presence of two types of adsorbed  $\text{H}_2\text{O}$  on the  $\text{TiO}_2$  surface. These were ascribed to chemisorbed and physisorbed  $\text{H}_2\text{O}$ , respectively, which was validated by variable-temperature  $^2\text{H}$  static NMR experiments (Fig. S10†). In the 1D  $^{17}\text{O}$  MAS NMR spectra (Fig. 1b), when the  $\text{H}_2\text{O}$  loading increases from 0.3 to  $4.7 \text{ mmol g}^{-1}$ , the  $^{17}\text{O}$  signal of adsorbed  $\text{H}_2\text{O}$  gradually narrows and shifts from -16.0 to 7.0 ppm, suggesting that the  $\text{H}_2\text{O}$  molecule is preferentially adsorbed on the unsaturated  $\text{Ti}_{5\text{C}}$  site to form chemisorbed  $\text{H}_2\text{O}$  ( $\text{Ti}_{5\text{C}}\text{-OH}_2$ ,  $^{17}\text{O}$  signal at -16 ppm), while excess  $\text{H}_2\text{O}$  molecules adsorb on the outer layer of chemisorbed  $\text{H}_2\text{O}$  and hydroxyls through hydrogen bonds to form physisorbed  $\text{H}_2\text{O}$  ( $^{17}\text{O}$  signal at 7.0 ppm).

Interestingly, the  $^1\text{H}$  signal (7.0 ppm) of bridging hydroxyl ( $\text{HO}_{2\text{C}}$ ) only correlates with the  $^{17}\text{O}$  signal of chemisorbed  $\text{H}_2\text{O}$  (-16.0 ppm); with the most possible structure illustrated in Fig. 2c  $M_{\text{I}}$ , while the  $^1\text{H}$  signal (1.7 ppm) of the terminal hydroxyl ( $\text{Ti}_{5\text{C}}\text{-OH}$ ) correlates with the  $^{17}\text{O}$  signals of both chemisorbed (Fig. 2c  $M_{\text{II}}$ ) and physisorbed  $\text{H}_2\text{O}$  (7.0 ppm, Fig. 2c  $M_{\text{III}}$ ). At the same time, as shown in Fig. 2b, there are two types of oxygen of hydroxyls present at 136 and 158 ppm, respectively. The  $^1\text{H}$  (7.0 ppm) signal of  $\text{HO}_{2\text{C}}$  is only correlated with the  $^{17}\text{O}$  signal at 158 ppm. As such, we assigned the cross

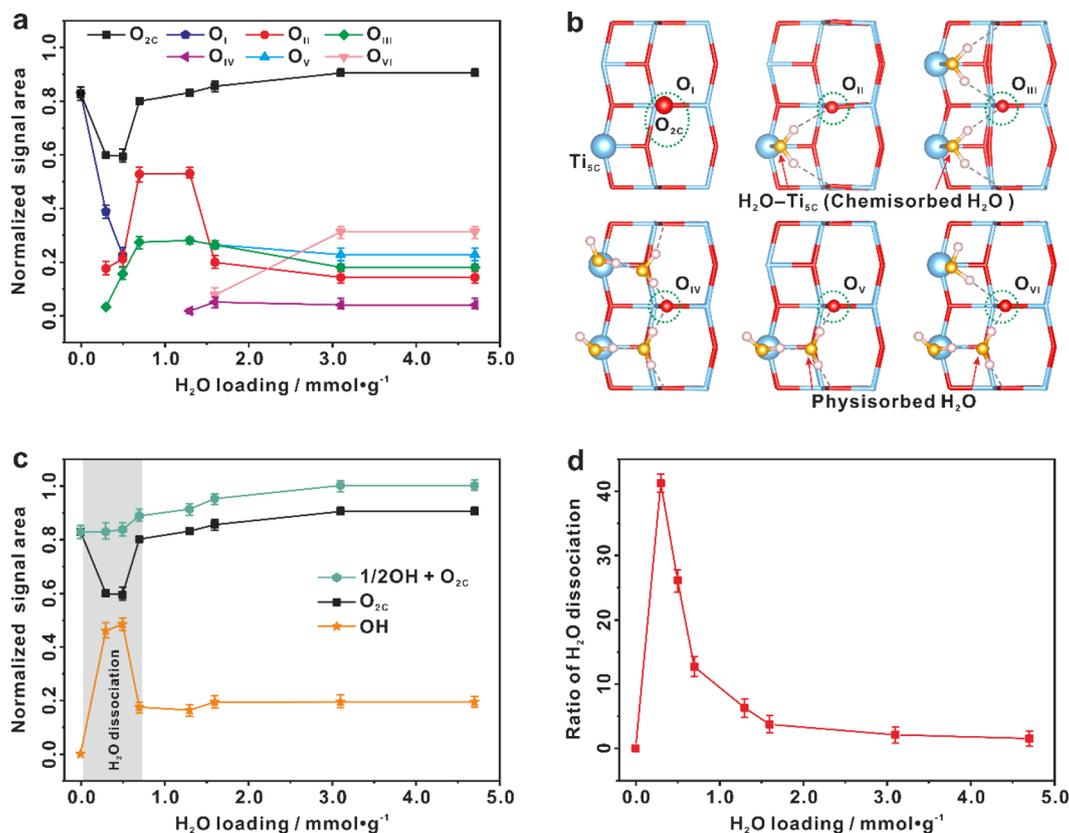
peak at (7.0, 158) ppm to the  $^1\text{H}$ - $^{17}\text{O}$  correlation from  $\text{HO}_{2\text{C}}$ , and ascribed the cross peak at (1.7, 136) ppm to the  $\text{Ti}_{5\text{C}}\text{-OH}$  correlation. The cross peak at (1.7, 158) ppm represents the correlation between them (Fig. 2c  $M_{\text{IV}}$ ). It is worth noting that the  $^1\text{H}$  signal of adsorbed  $\text{H}_2\text{O}$  (5.4 ppm) correlates with the  $^{17}\text{O}$  signals of both bridging OH (158 ppm, Fig. 2c  $M_{\text{V}}$ ) and terminal OH groups (136 ppm, Fig. 2c  $M_{\text{VI}}$ ), leading to two cross peaks at (5.4, 158) and (5.4, 136) ppm. As the  $\text{H}_2\text{O}$  loading increases to  $1.3 \text{ mmol g}^{-1}$ , the cross peaks become more prominent, but the interplay remains the same (Fig. S11†).

These results indicate that water splitting is easy to happen under ambient conditions over the practical  $\text{TiO}_2$  sample under conditions close to the working catalytic conditions, and more importantly, the interplay mode between the physisorbed or chemisorbed  $\text{H}_2\text{O}$  and the different types of hydroxyls is very complicated (Fig. 2c  $M_{\text{I}}$  to  $M_{\text{VI}}$ ), but it could be well resolved by NMR methods. However, questions remain to answer are, how the O-H bond of water ruptured over the  $\text{TiO}_2$  surface and whether the vacancy gets involved.

#### Quantitative evolution of surface oxygen/hydroxyl sites at the water/ $\text{TiO}_2$ interface during the hydration process

To answer the questions, quantitative information about different sites and species during water adsorption was estimated from 1D  $^{17}\text{O}$  MAS NMR spectra (Fig. 3a). As shown in Fig. 3a, the only surface  $\text{O}_{2\text{C}}$  site present on the bare  $\text{TiO}_2$





**Fig. 3** Spontaneous dissociation of interfacial H<sub>2</sub>O on the (101) facet of TiO<sub>2</sub> (100 mg) at room temperature. (a) Quantitative evolution of various surface two-coordinated oxygen sites (O<sub>2c</sub>, including O<sub>I</sub>–O<sub>VI</sub>) with the increase of H<sub>2</sub>O loading. (b) The possible configurations of six O<sub>2c</sub> sites (O<sub>I</sub>, O<sub>II</sub>, O<sub>III</sub>, O<sub>IV</sub>, O<sub>V</sub>, and O<sub>VI</sub>) on the (101) facet of TiO<sub>2</sub> optimized from theoretical calculations. (c) Quantitative evolution of surface O<sub>2c</sub> sites and hydroxyls with the increase of H<sub>2</sub>O loading. (d) The proportion of H<sub>2</sub>O dissociation with the increase of H<sub>2</sub>O loading. Quantification of the O<sub>2c</sub>/OH sites is conducted by fitting the main peaks and their spinning sidebands in the 1D <sup>17</sup>O MAS NMR spectra (Fig. 1b). Error bars in a, c, and d represent s.d. for each data point (three independent experiments), and points are the average values.

sample is O<sub>I</sub>. When the H<sub>2</sub>O loading is 0.3–0.5 mmol g<sup>-1</sup>, two new O<sub>2c</sub> sites (O<sub>II</sub>, O<sub>III</sub>) appear at the expense of the O<sub>I</sub> site. When the H<sub>2</sub>O loading grows up to 0.7 mmol g<sup>-1</sup>, the O<sub>II</sub> and O<sub>III</sub> sites reach maximum, while the O<sub>I</sub> site disappears completely. As shown in Fig. 1b, the chemisorbed H<sub>2</sub>O (Ti<sub>5c</sub>–OH<sub>2</sub>) is predominant in the H<sub>2</sub>O loading range of 0–0.7 mmol g<sup>-1</sup>. Therefore, the O<sub>II</sub> and O<sub>III</sub> structures should originate from the O<sub>I</sub> site interacting with chemisorbed H<sub>2</sub>O. With further increasing the H<sub>2</sub>O loading to 1.3 and then to 1.6 mmol g<sup>-1</sup>, the adsorbed H<sub>2</sub>O exists mainly in the form of physisorbed H<sub>2</sub>O (those without direct bonding with Ti sites), and meanwhile new O<sub>2c</sub> sites (O<sub>IV</sub>, O<sub>V</sub>, O<sub>VI</sub>) appear (Fig. 3a). In addition, the amount of O<sub>II</sub> and O<sub>III</sub> sites remains almost unchanged, implying that the physisorbed H<sub>2</sub>O molecules are mainly adsorbed on the outer-layer of chemisorbed H<sub>2</sub>O and OH groups rather than the surface O<sub>2c</sub> site. When the H<sub>2</sub>O loading is increased to 3.1 mmol g<sup>-1</sup>, the O<sub>VI</sub> site increases remarkably at the expense of O<sub>II</sub>, O<sub>III</sub>, and O<sub>V</sub> sites, indicating the further evolution of surface O<sub>2c</sub> sites interacting with the additional physisorbed H<sub>2</sub>O. The relative content of the O<sub>I</sub>–O<sub>VI</sub> sites eventually remains unchanged even when the H<sub>2</sub>O loading is further increased to 4.7 mmol g<sup>-1</sup>. With the above information,

we constructed six O<sub>2c</sub> sites with possible H<sub>2</sub>O adsorption patterns (Fig. 3b), which are derived from a series of calculated structures with different H<sub>2</sub>O adsorption configurations on (2 × 2) surface slabs (Fig. S12–S24<sup>†</sup>). Obviously, the calculated <sup>17</sup>O NMR chemical shifts of these O<sub>2c</sub> sites just correspond to experimental observation (Fig. S25<sup>†</sup>).

The sum of O<sub>2c</sub> (O<sub>2c</sub> = O<sub>I</sub> + O<sub>II</sub> + O<sub>III</sub> + O<sub>IV</sub> + O<sub>V</sub> + O<sub>VI</sub>) sites shows quantitative information about the surface O<sub>2c</sub> sites (Fig. 3a and c). It is relatively constant except at the water loading of 0.3–0.5 mmol g<sup>-1</sup>, where the total content of surface O<sub>2c</sub> sites is decreased by ca. 20% (see the grey region highlighted in Fig. 3c). Very interestingly, this is just the time for the appearance of OH groups whose content is increased by ca. 40% (Fig. 3c). This, together with the above results (Fig. 2), shows that the chemisorbed H<sub>2</sub>O molecule on the Ti<sub>5c</sub> site (H<sub>2</sub>O–Ti<sub>5c</sub>) is apt to dissociate into a terminal OH group (Ti<sub>5c</sub>–OH) and a proton (H<sup>+</sup>), and the latter protonates an adjacent surface O<sub>2c</sub> site to form a bridging OH group (HO<sub>2c</sub>), which “consumes” an adjacent O<sub>2c</sub> site (see below Fig. 4). The proportion of water dissociation relative to total water loaded was also calculated, as depicted in Fig. 3d. Intriguingly, the highest proportion of H<sub>2</sub>O dissociation, at 41.2%, was found at a water loading of 0.3 mmol





Fig. 4 Schematic diagram of H<sub>2</sub>O dissociation and regeneration on the TiO<sub>2</sub> surface.

$\text{g}^{-1}$ . With increasing the H<sub>2</sub>O loading from 0.5 to 4.7  $\text{mmol g}^{-1}$ , the content of surface O<sub>2C</sub> sites gradually recovered and that of the hydroxyls decreased (Fig. 3c). The results point to the reaction of HO<sub>2C</sub> groups with nearby Ti<sub>5C</sub>-OH groups, leading to the regeneration of H<sub>2</sub>O-Ti<sub>5C</sub> and the recovery of surface O<sub>2C</sub> sites. The schematic diagram of H<sub>2</sub>O dissociation and regeneration on the TiO<sub>2</sub> surface is illustrated in Fig. 4.

The dissociation process of interfacial H<sub>2</sub>O could be also confirmed by 1D <sup>2</sup>H MAS NMR experiments on the TiO<sub>2</sub> samples (Fig. 5 and S26†). The dehydrated <sup>2</sup>H-enriched TiO<sub>2</sub> was prepared by exchanging TiO<sub>2</sub> with 4.7  $\text{mmol g}^{-1}$  of <sup>2</sup>H<sub>2</sub>O at room temperature for 2 h, and then the <sup>2</sup>H-enriched sample was dehydrated at 160 °C and then loaded with different amounts (0.3–4.7  $\text{mmol g}^{-1}$ ) of <sup>2</sup>H<sub>2</sub>O at room temperature. Since the proton of the bridging OH group is readily exchanged with <sup>2</sup>H of D<sub>2</sub>O, only the <sup>2</sup>H signal of the deuterated bridging OH group (DO<sub>2C</sub>) is observable at 7.4 ppm in the 1D spectrum of dehydrated TiO<sub>2</sub> (Fig. 5a). When 0.3  $\text{mmol g}^{-1}$  D<sub>2</sub>O is adsorbed on the deuterated TiO<sub>2</sub>, in addition to the increase of the deuterated bridging OH group, a new signal at 1.6 ppm due to the deuterated terminal OH group (Ti<sub>5C</sub>-OD) appears, which originates from the dissociation of D<sub>2</sub>O (Fig. 5a). The broad signal at 4.4 ppm corresponds to adsorbed D<sub>2</sub>O. With the increase of D<sub>2</sub>O loading, there is a similar evolution trend for the DO<sub>2C</sub> and Ti<sub>5C</sub>-OD groups, especially their signal increment is roughly equal (Fig. 5b), consistent with the theoretical content ratio (1 : 1) of the two types of deuterated hydroxyls originating from the D<sub>2</sub>O dissociation. As shown in Fig. 5b, the content of DO<sub>2C</sub> and Ti<sub>5C</sub>-OD groups on the TiO<sub>2</sub> surface reaches maximum at a 0.5  $\text{mmol g}^{-1}$  H<sub>2</sub>O loading, in line with the result of 1D <sup>17</sup>O MAS NMR analysis (Fig. 3c).

## Theoretical insight into the adsorption and dissociation of interfacial H<sub>2</sub>O

Based on our NMR results, a series of structures with different H<sub>2</sub>O adsorption configurations on (2 × 2) surface slabs are optimized (Fig. S12–S24†) by density functional theory (DFT) calculations. From the adsorption energy points of view, it can be found that when the same amount of H<sub>2</sub>O is adsorbed onto the TiO<sub>2</sub> (101) (2 × 2) surface slabs, the more the chemisorbed H<sub>2</sub>O (Ti<sub>5C</sub>-OH<sub>2</sub>), the greater the adsorption energy of H<sub>2</sub>O will be released (Table S2†), indicating that H<sub>2</sub>O is preferentially chemisorbed on the Ti<sub>5C</sub> site, forming Ti<sub>5C</sub>-OH<sub>2</sub>, and then excess H<sub>2</sub>O is physisorbed on the Ti<sub>5C</sub>-OH<sub>2</sub> site through the hydrogen bond, consistent with our NMR experimental results. Thus, with the increase of H<sub>2</sub>O loading, the constructed 1H<sub>2</sub>O/1Ti<sub>5C</sub>, 2H<sub>2</sub>O/2Ti<sub>5C</sub>, 3H<sub>2</sub>O/3Ti<sub>5C</sub>, 4H<sub>2</sub>O/4Ti<sub>5C</sub>, 5H<sub>2</sub>O/4Ti<sub>5C</sub>, 6H<sub>2</sub>O/4Ti<sub>5C</sub>, and 8H<sub>2</sub>O/4Ti<sub>5C</sub> configurations are energy optimal and appear gradually on the TiO<sub>2</sub> (101) surface with the surface O<sub>2C</sub> sites consisting of the O<sub>I</sub>-O<sub>VI</sub> sites (Fig. 6a).

Our DFT calculations revealed that water dissociation is a coverage-dependent process. Increasing the H<sub>2</sub>O loading on the Ti<sub>5C</sub> site of (2 × 2) surface slabs from 1H<sub>2</sub>O/1Ti<sub>5C</sub> to 4H<sub>2</sub>O/4Ti<sub>5C</sub>, the H-O-H bond angle of chemisorbed H<sub>2</sub>O is gradually twisted (Fig. 6b), leading to a decline of the dissociation energy of chemisorbed H<sub>2</sub>O (Fig. 6c). At the optimal water coverage of 4H<sub>2</sub>O per TiO<sub>2</sub> (101) (2 × 2) surface slab (4H<sub>2</sub>O/4Ti<sub>5C</sub>), water splitting is enhanced, with the dissociation energy dropping to as low as −0.03 eV. This suggests that H<sub>2</sub>O dissociation is both exothermic and spontaneous, aligning with our experimental findings. Obviously, the more chemisorbed H<sub>2</sub>O on the localized (101) facet of TiO<sub>2</sub>, the more favorable to the dissociation of H<sub>2</sub>O to form OH. With the further increase of the H<sub>2</sub>O loading on the Ti<sub>5C</sub> site of (2 × 2) surface slabs (5H<sub>2</sub>O/4Ti<sub>5C</sub>, 6H<sub>2</sub>O/4Ti<sub>5C</sub>, and 8H<sub>2</sub>O/4Ti<sub>5C</sub>), the presence of physisorbed H<sub>2</sub>O causes a recovery in the average bond angle of H<sub>2</sub>O (Fig. 6b) and an increase in dissociation energy (Fig. 6c), implying that physisorbed H<sub>2</sub>O impedes the dissociation of chemisorbed H<sub>2</sub>O.

To further address the importance of the interplay between OH and H<sub>2</sub>O, we also computed the optimized configurations of

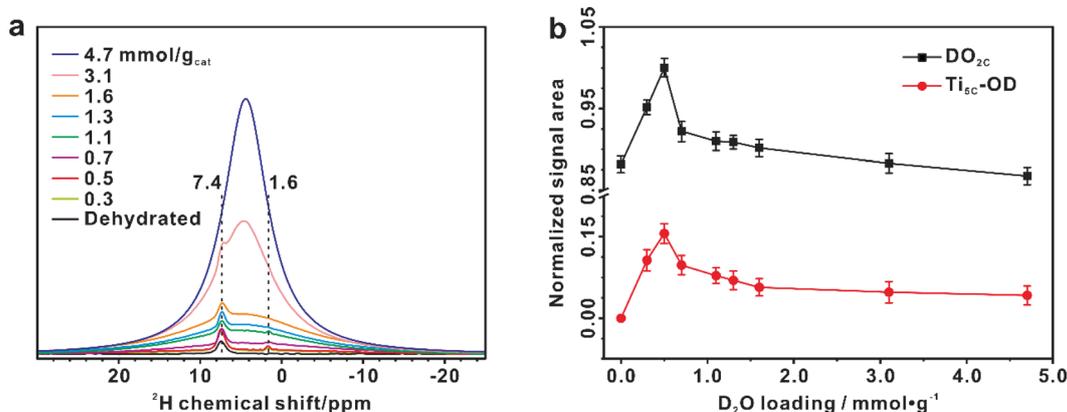


Fig. 5 Dissociation of interfacial H<sub>2</sub>O validated by 1D <sup>2</sup>H MAS NMR spectra. (a) <sup>2</sup>H MAS NMR spectra of dehydrated <sup>2</sup>H-enriched TiO<sub>2</sub> samples with <sup>2</sup>H<sub>2</sub>O loading from 0 to 4.7  $\text{mmol g}^{-1}$ . (b) Quantitative evolution of the deuterated bridging OH (DO<sub>2C</sub>, at 7.4 ppm) and terminal OH (Ti<sub>5C</sub>-OD, at 1.6 ppm) groups with the increase of H<sub>2</sub>O loading, derived from the simulation of the 1D <sup>2</sup>H MAS NMR spectra (Fig. S26†).



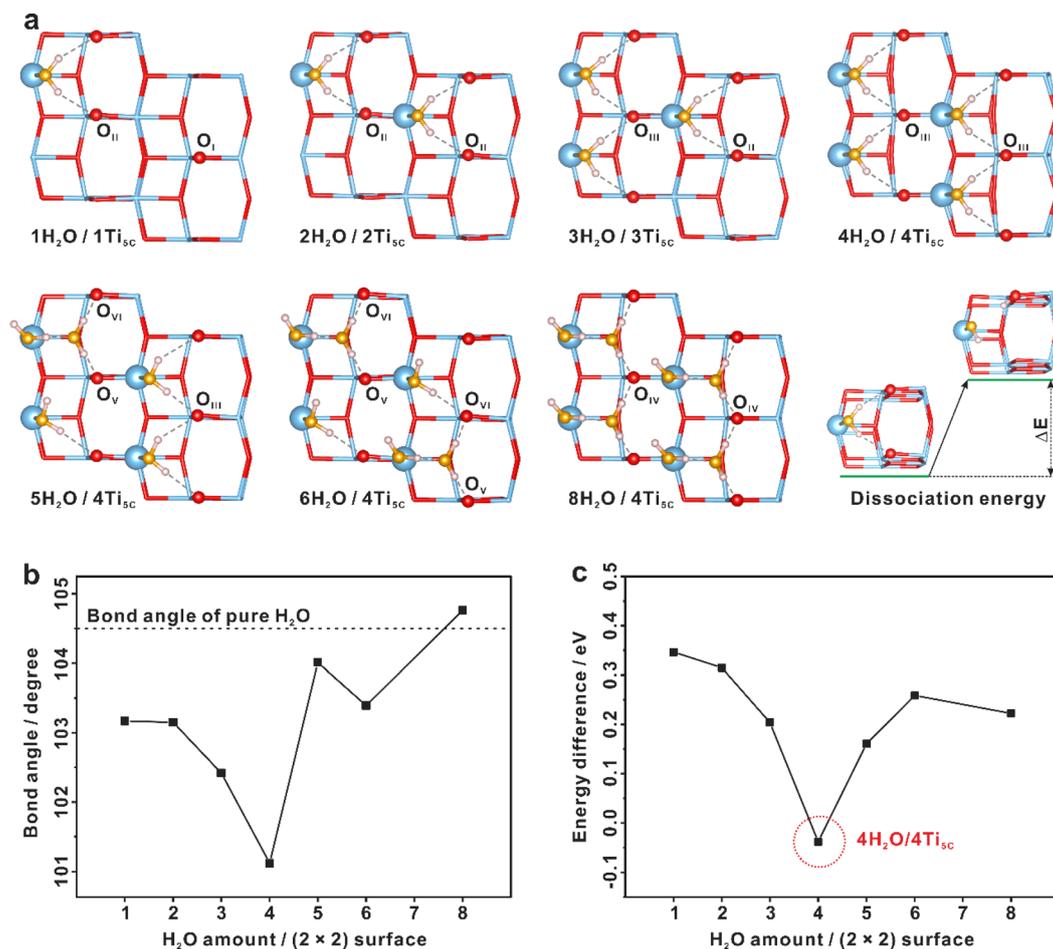


Fig. 6 Dissociation of interfacial H<sub>2</sub>O on the TiO<sub>2</sub> surface revealed by DFT calculations. (a) Calculated structures of the TiO<sub>2</sub> (101) (2 × 2) surface slabs with different H<sub>2</sub>O adsorption configurations. Titanium and oxygen atoms of TiO<sub>2</sub> are plotted in blue (Ti) and red (O), oxygen atoms originating from the adsorbed H<sub>2</sub>O are plotted in yellow (H<sub>2</sub>O), and hydrogen atoms are plotted in the pink (H). (b) The average bond angle of interfacial H<sub>2</sub>O on the (2 × 2) surface slabs with different H<sub>2</sub>O adsorption configurations, calculated from Table S3.† (c) Dissociation energy of interfacial H<sub>2</sub>O in the optimized structures of different H<sub>2</sub>O adsorption configurations on (2 × 2) TiO<sub>2</sub> (101) surface slabs.

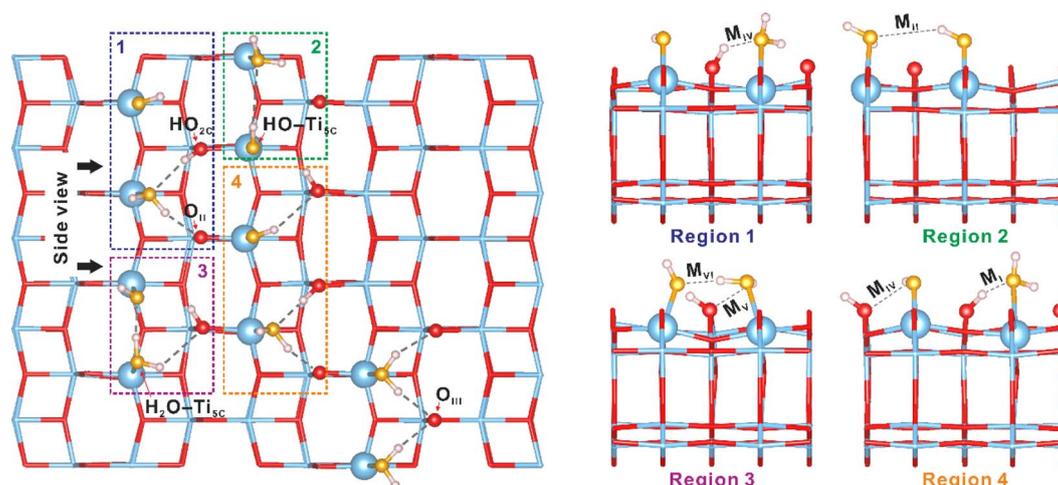


Fig. 7 The role of the OH–H<sub>2</sub>O interactions in the dissociation of interfacial H<sub>2</sub>O revealed by DFT calculations. The interaction modes of the hydroxyls with chemisorbed H<sub>2</sub>O on the (101) facet of TiO<sub>2</sub> at a 40% coverage of water by using a large (5 × 5 × 1) supercell slab with 300 atoms (100 Ti atoms and 200 O atoms), which highlights the optimized atomic structures (M<sub>I</sub>–M<sub>IV</sub>) of hydroxyls and adsorbed H<sub>2</sub>O molecules on the surface of TiO<sub>2</sub> shown in Fig. 2c. Regions 1–4 show side views of those structures (M<sub>I</sub>–M<sub>IV</sub>) in the 1–4 regions. Titanium and oxygen atoms of TiO<sub>2</sub> are plotted in blue (Ti) and red (O), oxygen atoms originating from the adsorbed H<sub>2</sub>O are plotted in yellow (H<sub>2</sub>O), and hydrogen atoms are plotted in pink (H).



the H<sub>2</sub>O dissociation state on the (101) facet of TiO<sub>2</sub> at 40% coverage by using a large (5 × 5 × 1) supercell slab with 300 atoms (100 Ti atoms and 200 O atoms), as presented in Fig. 7 and S27.† It's evident that the OH groups originating from water dissociation have strong interactions with the neighboring chemisorbed H<sub>2</sub>O in the M<sub>I</sub>–M<sub>VI</sub> conformations, corroborating with our 2D <sup>1</sup>H{<sup>17</sup>O} J-HMQC NMR findings (Fig. 2). This robust interaction stabilizes H<sub>2</sub>O dissociation, serving as the driving force for the event. Therefore, at water loadings ranging from 0.3 to 0.5 mmol g<sup>-1</sup> (approximately 28–46% coverage), massive H<sub>2</sub>O dissociation takes place. At elevated water loadings, physisorbed H<sub>2</sub>O molecules appear, potentially disrupting the interaction between bridging OH groups (Ti–HO<sub>2C</sub>–Ti) and chemisorbed H<sub>2</sub>O. This promotes the reaction of bridging OH groups with nearby terminal OH groups, leading to the reformation of molecular H<sub>2</sub>O (Fig. S28†). All these findings underscore the crucial role of the interplay between OH–H<sub>2</sub>O and H<sub>2</sub>O–H<sub>2</sub>O interactions in determining the behavior of interfacial H<sub>2</sub>O on the TiO<sub>2</sub> surface.

## Conclusions

In summary, the atomic-level structures and quantitative evolution of surface O/OH sites along with H<sub>2</sub>O loading are identified on the TiO<sub>2</sub> (101) surface by solid-state NMR spectroscopy coupled with theoretical calculations. Two types of H<sub>2</sub>O (chemisorbed H<sub>2</sub>O, *i.e.* Ti<sub>5C</sub>–OH<sub>2</sub> and physisorbed H<sub>2</sub>O) are present on the H<sub>2</sub>O/TiO<sub>2</sub> interface, and detailed interactions between the interfacial H<sub>2</sub>O and surface O<sub>2C</sub>/OH sites are ascertained. We need to mention that the results of the quantitative <sup>17</sup>O NMR and <sup>2</sup>H NMR tests have confirmed that, under ambient conditions, a maximum of 41.2% of the total adsorbed H<sub>2</sub>O on TiO<sub>2</sub> spontaneously dissociates at a water loading of 0.3 mmol g<sup>-1</sup>. This means that approximately 0.124 mmol g<sup>-1</sup> of water dissociates over TiO<sub>2</sub> under these conditions. Our NMR findings and DFT calculations, together with ESR results, confirm that the responsible factor for the observed water rupture at room temperature over the TiO<sub>2</sub> (101) surface is not defects such as oxygen vacancies and impurities, but the joint effect of the coordination-unsaturated Ti<sub>5C</sub> site and adjacent O<sub>2C</sub> site, as well as the delicate interplay between water–surface and water–water interactions.

## Data availability

The data that support the findings of this study are available within the article and the ESI.†

## Author contributions

F. D., N. F., and D. M. conceived the project. N. F. designed the studies. L. Y. synthesized the TiO<sub>2</sub> photocatalysts. N. F., L. Y., J. X., M. W. performed NMR experiments. M. W., Y. J., D. M., L. Y., N. F. and F. D. analyzed all the experimental data. M. H. performed theoretical calculations. N. F., D. M. and F. D. wrote the manuscript. All authors interpreted the data and contributed to the preparation of the manuscript.

## Conflicts of interest

The authors declare no competing interests.

## Acknowledgements

This work was supported by the National Natural Science Foundation of China (no. 22372177, 22127801, 22225205, and 22320102002), Strategic Priority Research Program of the Chinese Academy of Sciences (XDB0540000), Natural Science Foundation of Hubei Province (2021CFA021), Hubei International Scientific and Technological Cooperation program (2022EHB021), and International Science & Technology Cooperation Base for Sustainable Catalysis and Magnetic Resonance (SH2303). This work has been supported by the New Cornerstone Science Foundation. D. M. acknowledges support from the Tencent Foundation through the XPLOER PRIZE.

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