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# A scalable approach using a gC<sub>3</sub>N<sub>4</sub>-covalent organic framework hybrid catalyst towards sustainable hydrogen production from seawater and wastewater†

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The photocatalytic generation of H<sub>2</sub> using covalent organic frameworks (COFs) is gaining more interest. While numerous reports have focused on the production of H<sub>2</sub> from deionized water using COFs, the inability to produce H<sub>2</sub> from industrial wastewater or seawater is a common limitation in many reported catalysts. Additionally, many of these reports lack a clear path to scale up the catalyst synthesis. In this study, we explore the prospect of hybridizing a COF with gC<sub>3</sub>N<sub>4</sub> to create a robust photocatalyst for efficient H<sub>2</sub> generation. This hybrid exhibits outstanding performance not only in deionized water, but also in wastewater, and simulated seawater. Furthermore, we explore the feasibility of the bulk-scale synthesis and successfully produce a 20 g hybrid catalyst in a single batch, and the synthesis method is scalable to achieve the commercial target. Remarkably, a maximum HER rate of 94 873 μmol g<sup>-1</sup> h<sup>-1</sup> and 109 125 μmol g<sup>-1</sup> h<sup>-1</sup> was obtained for the hybrid catalyst from industrial wastewater and simulated seawater, respectively. The performance of bulk-scale batches closely matches that of the small-scale ones. This research paves the way for the utilization of organic photocatalysts on a commercial scale, offering a promising solution for sustainable large-scale H<sub>2</sub> production.

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## Introduction

The incorporation of renewable resources in energy generation is an essential pillar for a sustainable future. Among the potential alternatives, hydrogen (H<sub>2</sub>), the most prevalent molecule in the universe, stands out as a clean fuel with the potential to supplant conventional fossil fuels. This transition to H<sub>2</sub> is driven by advantages including abundance, emission-free combustion, high energy efficiency, and inherent renewability.<sup>1</sup> Nevertheless, the practical adoption of H<sub>2</sub> encounters certain constraints, particularly concerning long-distance transport and extended-term storage. Issues such as flammability pose challenges in the entire supply and demand chain,

and the production of green H<sub>2</sub> at a large scale may entail elevated costs.<sup>1</sup> The conventional methods for H<sub>2</sub> production involve sources such as natural gases, naphtha, heavy oil, coal, and electrolysis, and industrial-scale production primarily relies on the Bosch process.<sup>2</sup> However, when exploring sustainable avenues for producing green H<sub>2</sub>, one cannot overlook the fact that 71% of our planet is covered with water, making it an abundant source of H<sub>2</sub>. By harnessing solar energy and employing catalytic processes to extract H<sub>2</sub> from water, we have the opportunity to create a bountiful, clean energy source that aligns with the goals of decarbonization.

Over the past decade, organic photocatalysts have gained increasing prominence in the field of water splitting.<sup>3–5</sup> In 2014, Lotsch and co-workers reported, for the first time, photocatalytic generation of H<sub>2</sub> (1970 μmol g<sup>-1</sup> h<sup>-1</sup>) from water using a hydrazone-based covalent organic framework (COF) as the photosensitizer.<sup>6</sup> Subsequently, numerous research groups have explored the use of various COFs for the photocatalytic generation of H<sub>2</sub> from water.<sup>7,8</sup> Over time, several strategies have been employed to enhance the stability of photocatalysts and improve H<sub>2</sub> evolution efficiency. They include molecular engineering,<sup>9,10</sup> multivariate synthesis,<sup>11</sup> defect engineering,<sup>12</sup> dye sensitization,<sup>13</sup> electron transfer mediators,<sup>14</sup> hybridizations,<sup>15–23</sup> etc. These efforts have led to significant

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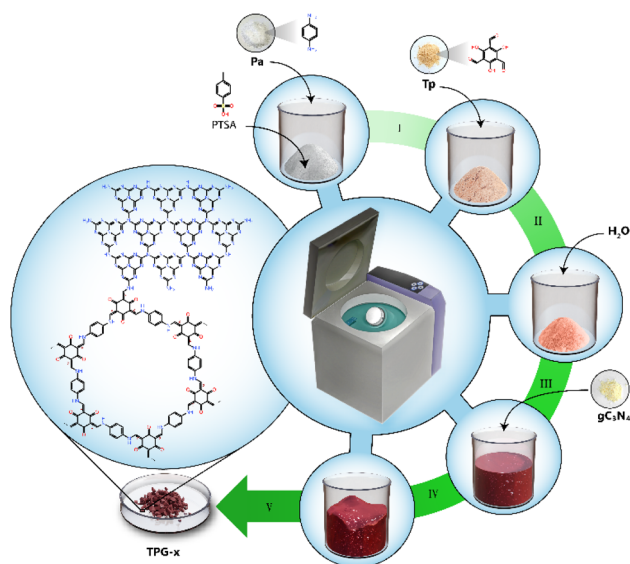
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**Scheme 1** Bulk scale synthesis strategy adopted for the preparation of TPG-*x* photocatalysts. Step I: mix PTSA and Pa in a planetary mixer. Step II: add Tp and mix again. Step III: add an adequate amount of H<sub>2</sub>O to the reaction mixture and mix. Step IV: add gC<sub>3</sub>N<sub>4</sub> to the reaction mixture and mix. Step V: heat the reaction mixture in an oven at 90 °C, followed by washing and drying to yield the final catalyst in powder form.

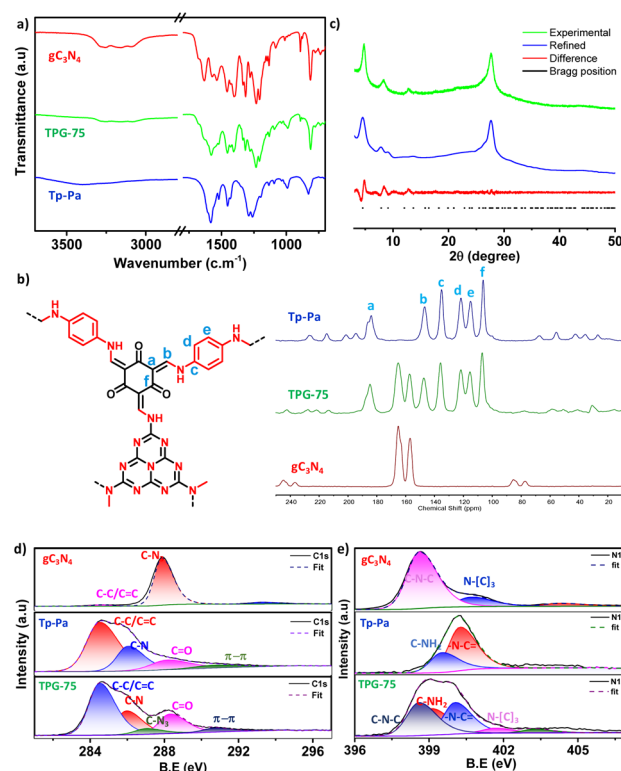
advancements in the field, demonstrating the versatility and potential of COFs in sustainable H<sub>2</sub> production. Hybridization of COFs using metallic conductors, MOFs, inorganic semiconductors, and polymers to form heterojunctions within COFs has recently emerged as a strategy for photocatalytic H<sub>2</sub> evolution (PHE).<sup>24</sup> Even though COF hybridization with gC<sub>3</sub>N<sub>4</sub> was reported earlier, the synthesis procedure follows the solvothermal method which limits the opportunity for bulk-scale synthesis.<sup>25,26</sup> Moreover, the photocatalytic performance was tested in deionized water and the H<sub>2</sub> evolution efficiency was not up to the mark. Ming *et al.* reported a metal-insulator-semiconductor-based photosystem comprising Tp-COF with polyvinylpyrrolidone (PVP) coated Pt nanoparticles (NPs) for the photocatalytic production of H<sub>2</sub>.<sup>27</sup> Unlike bare Pt NPs, here the photoexcited  $\pi$ -electrons in the n-type Tp-COF semiconductor were effectively extracted and tunnelled towards Pt NPs through an ultrathin PVP insulating layer which further enhanced the activity. While significant progress has been made in PHE, most high-performing photocatalysts have been tested using deionized water for H<sub>2</sub> production.<sup>28–45</sup> It's important to note that only a small fraction of Earth's water resources, about 1%, is fresh water. Therefore, exploring PHE from alternative sources such as seawater, industrial wastewater, or non-potable water is of great importance. However, there are limited reports on H<sub>2</sub> generation from seawater using COFs due to the challenges posed by side reactions occurring on the catalyst surface and thereby reducing the efficiency of photogenerated charge carriers in photocatalytic processes.<sup>46–48</sup>

However, in 2023, Yue *et al.* reported a significant breakthrough with a reasonably high-performance H<sub>2</sub> production

rate of 41 300  $\mu\text{mol g}^{-1} \text{h}^{-1}$  using **Tp-Pa**, outperforming many other organic, inorganic, and hybrid materials employed for PHE from seawater under visible light irradiation.<sup>49</sup> However, the absence of reports addressing sustainable H<sub>2</sub> production using scalable photocatalysts, despite numerous studies demonstrating high rates of H<sub>2</sub> evolution using various synthetic methodologies, is a significant gap in the research field. Herein, we introduce a hybrid catalyst comprising **Tp-Pa** and gC<sub>3</sub>N<sub>4</sub> capable of reasonably high H<sub>2</sub> evolution from both seawater and industrial wastewater. In this work, we aim to tackle two major challenges: (1) H<sub>2</sub> production from industrial wastewater and seawater, and (2) bulk-scale synthesis of the photocatalyst (Scheme 1). These efforts not only expand the application of COFs in environmentally significant areas but also make the technology more accessible for practical, large-scale use.

## Results and discussion

All the COFs including hybrid COFs **Tp-Pa-gC<sub>3</sub>N<sub>4</sub>-*x*s** (TPG-*x*, *x* is the percentage of gC<sub>3</sub>N<sub>4</sub> to the total quantity of Tp and Pa and it varies as 10, 25, 50, 75, and 100) were prepared using mechanochemical synthesis on a 2 g scale (Schemes S1† and 1, and Table S1†).<sup>50</sup> A representative chemical structure of TPG-*x* is shown in Scheme 1. All the COFs were characterized by FT-IR and intense peaks at around  $\sim 1226 \text{ cm}^{-1}$  (–C–N) and  $\sim 1551 \text{ cm}^{-1}$  (–C=C) revealed the formation of a  $\beta$ -ketoenamine linked framework for both **Tp-Pa** and TPG-*x*s (Fig. 1 and S1†).



**Fig. 1** Comparison of (a) FT-IR, (b) <sup>13</sup>C CP MAS NMR, (c) PXRD and XPS for (d) C 1s, and (e) N 1s of **Tp-Pa**, gC<sub>3</sub>N<sub>4</sub> and TPG-75.



Also, the stretching frequencies of nitrogen-containing heterocycles were evident on both  $\text{gC}_3\text{N}_4$  and **TPG-x**s. A broad peak at around  $\sim 3250\text{ cm}^{-1}$  was obtained for  $\text{gC}_3\text{N}_4$ , which corresponds to the stretching frequency of N–H bonds. Furthermore, the intensity of N–H stretching frequency was reduced for **TPG-x**s, thereby confirming the bond formation between the free aldehyde group of **Tp-Pa** and  $\text{NH}_2$  of  $\text{gC}_3\text{N}_4$ , which is in line with the previous literature report.<sup>27,51</sup> Solid-state  $^{13}\text{C}$  cross-polarization magic angle spinning (CP-MAS) NMR spectroscopy was utilized to provide evidence of hybrid COF formation. The peaks at around 104.6 and 182 ppm for **Tp-Pa** and **TPG-x** correspond to  $-\text{C}=\text{O}$  and  $-\text{C}=\text{C}-$  bond formation, respectively; this provided critical insights into the composition and structural characteristics of the synthesized materials, further validating hybrid COF formation and enhancing our understanding of these novel materials. At the same time, pristine  $\text{gC}_3\text{N}_4$  shows peaks at around 156 and 165 ppm, corresponding to the C–N and C–C bonds of heptazine units. As evident from the graph, hybrid COFs show a gradual increase in the intensity of peaks (heptazine unit) from 0 to 100% of  $\text{gC}_3\text{N}_4$  in **TPG-x** (Fig. 1 and S2†). The crystallinity of the samples was analyzed using powder X-ray diffraction (PXRD). All **TPG-x**s show a sharp crystalline peak around  $2\theta$  of 4.8, corresponding to reflection from the (100) plane (Fig. 1 and S3†). Similarly, small intense peaks at around 8.07 and 12.9 were also observed, indicating reflection from the (200), and (210) planes, respectively. Apart from this, another peak at 27.5 corresponding to reflection from the (001) plane is also visible, which confirms the  $\pi$ – $\pi$  stacking arising in the COF.  $\text{gC}_3\text{N}_4$  shows one sharp intense peak at around  $2\theta \approx 27.6$  arising from the stacking of the conjugated heptazine ring.<sup>52</sup> Notably, the intensity of the peak from the (001) plane increases gradually as the ratio of  $\text{gC}_3\text{N}_4$  increases because of its overlap with the peak from the (001) plane of **Tp-Pa**.<sup>53</sup> Thermogravimetric analysis (TGA) of COFs shows thermal stability up to 400 °C corresponding to a robust framework. All samples of **TPG-x** showed two major degradation peaks, at around 400 °C corresponding to **Tp-Pa** and at around 600 °C corresponding to  $\text{gC}_3\text{N}_4$  (Fig. 1 and S4†). The decomposition of all the samples in air points to the composition of the samples.

As a representative example, the detailed characterization of **TPG-75** is provided in Fig. 1 and 2. X-ray photoelectron spectroscopy (XPS) revealed the presence of C 1s, N 1s, and O 1s in both pristine and hybrid COFs (Fig. 1 and S5†). Furthermore, the C 1s spectrum of  $\text{gC}_3\text{N}_4$ , **Tp-Pa**, and **TPG-75** was deconvoluted into different peaks.<sup>27,51</sup> As shown in Fig. 1d, the C 1s spectrum of pristine  $\text{gC}_3\text{N}_4$  can be deconvoluted into one major individual peak at 287.9 eV corresponding to the C–N bond along with an adventitious carbon peak at 284.6 eV. At the same time, the spectrum of the hybrid COF can be deconvoluted into 4 individual peaks at 284.6, 285.9, 288.3, and 290.7 corresponding to C–C/C=C, C–N, C=O, and  $\pi$ – $\pi$  interaction, respectively. Similarly, a full survey of the N 1s spectrum reveals the individual N1s peaks of both  $\text{gC}_3\text{N}_4$  and **Tp-Pa** in **TPG-75** (Fig. 1e). Hence XPS of **TPG-75** shows a combination of  $\text{gC}_3\text{N}_4$  and **Tp-Pa**. The morphology of the samples was analyzed using Field Emission Scanning Electron Microscopy (FE-SEM) and High-Resolution Transmission Electron Microscopy (HR-TEM)

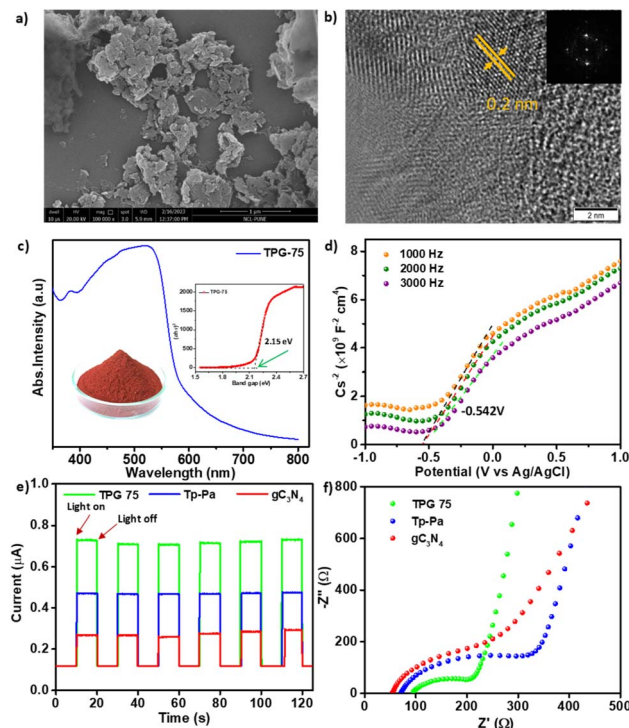


Fig. 2 (a) FE-SEM image of **TPG-75**. (b) HR-TEM image of **TPG-75** showing lattice fringes and  $d$ -spacing; SAED pattern is shown as an inset. (c) UV-vis DRS pattern of **TPG-75** (inset shows the corresponding Tauc plot). (d) Mott–Schottky measurement of **TPG-75**. (e) Photo-current measurements of **TPG-75**, **Tp-Pa**, and  $\text{gC}_3\text{N}_4$  under visible light irradiation on and off conditions with a time interval of 20 seconds. (f) EIS of **TPG-75**, **Tp-Pa**, and  $\text{gC}_3\text{N}_4$ .

(Fig. 2a, b and S6–S12†).  $\text{gC}_3\text{N}_4$  showed a sheet-like morphology with a thickness of around 20–30 nm and stacking of sheets to form a layered structure (Fig. S7†). HR-TEM images also confirmed the sheet-like morphology of **TPG-75** having crystalline fringes with a  $d$ -spacing of 0.2 nm (Fig. 2b). The selected area electron diffraction (SAED) pattern also confirmed the crystallinity of the samples at a lower amount of  $\text{gC}_3\text{N}_4$ . As the percentage increased, the fringes were not visible, especially for **TPG-100** (Fig. S12†). At the same time, the characteristic morphology of  $\text{gC}_3\text{N}_4$  was more evident in **TPG-75** and **TPG-100** (Fig. S11 and S12†). Elemental mapping using Scanning Tunneling Electron Microscopy (STEM) was also carried out and it confirmed the presence of C (81%), N (15%), and O (4%) in **TPG-75** (Fig. S13†).

The permanent porosity and surface area of the samples were analyzed using Brunauer–Emmett–Teller (BET) and Density Functional Theory (DFT) pore size distribution methods. **TPG-x**s showed a lower surface area compared to pristine **Tp-Pa** due to the incorporation of  $\text{gC}_3\text{N}_4$ . A BET surface area of  $652\text{ m}^2\text{ g}^{-1}$  was obtained for **TPG-75** which is less compared to the surface area of **Tp-Pa** ( $1171\text{ m}^2\text{ g}^{-1}$ ). It is to be noted that  $\text{gC}_3\text{N}_4$  exhibited a very low surface area of  $87\text{ m}^2\text{ g}^{-1}$  and increasing  $\text{gC}_3\text{N}_4$  content gradually decreased the surface area of **TPG-x**s (Fig. S14–S20†). Ultraviolet-visible diffuse reflectance spectroscopy (UV-vis DRS) was used to measure the





absorption features and all COFs exhibited a broad absorption ranging from 300 to 750 nm whereas the absorption of  $\text{gC}_3\text{N}_4$  is limited to 300 to 400 nm (Fig. S21†). The optical energy band gaps were calculated from the Tauc plot for all catalysts and it was found that  $\text{gC}_3\text{N}_4$  has a broader band gap of 3.1 eV whereas **Tp-Pa** exhibits a narrow band gap of 2.14 eV (Fig. 2c). The obtained band gap values are comparable with previously reported values.<sup>27,51</sup> The band gap of **TPG-75** was 2.15 eV, which is similar to the band gap of **Tp-Pa** (Fig. S22†). At the same time, the Mott-Schottky measurements exhibit a positive slope denoting n-type semiconductor characteristics for the hybrid catalyst (Fig. 2d), pristine  $\text{gC}_3\text{N}_4$ , and **Tp-Pa** (Fig. S23†). In order to understand the light responses, the photocurrent measurement of **TPG-75** was conducted and compared with that of **Tp-Pa** and  $\text{gC}_3\text{N}_4$  (Fig. 2e). **TPG-75** shows a higher photocurrent response compared to **Tp-Pa** and  $\text{gC}_3\text{N}_4$  during several on-off photo-irradiation cycles under identical experimental conditions. A sequential increment in photoresponse was noticed from **Tp-Pa** to  $\text{gC}_3\text{N}_4$  and further to **TPG-75**. Furthermore, the results from the electrochemical impedance spectra (EIS) reveal a notable reduction in charge transfer resistance for charge separation in **TPG-75** compared to that of **Tp-Pa** and  $\text{gC}_3\text{N}_4$  (Fig. 2f). These findings collectively affirm that the hybridization process has brought about a substantial enhancement in charge separation within the COF platform.

PHE of all hybrid COFs, **Tp-Pa**, and  $\text{gC}_3\text{N}_4$  was carried out in a quartz round bottom flask of 50 mL capacity under visible light irradiation in the presence of SED and a cocatalyst. In a typical experiment, 5 mg of catalyst was dispersed in an aqueous solution of 0.056 M (200 mg) ascorbic acid (AA) in 20 mL  $\text{H}_2\text{O}$ . PVP-coated Pt NPs were synthesized according to the reported literature<sup>54</sup> and used as such as the cocatalyst. The HER rate of **Tp-Pa** was found to be  $26\,126\ \mu\text{mol g}^{-1}\text{ h}^{-1}$ , which is comparable with that in the previous report (Fig. 3a).<sup>49</sup> It has to be noted that  $\text{gC}_3\text{N}_4$  showed no activity with AA as SED under

identical experimental conditions (Fig. 3a). Surprisingly, hybrid catalysts with different percentages of  $\text{gC}_3\text{N}_4$  performed much better than **Tp-Pa**, delivering a nearly three-fold increase in activity. For instance, **TPG-75** showed a remarkably high catalytic activity of  $85\,617$  and  $179\,064\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  under visible light and simulated solar light illumination, respectively (Fig. 3b). An increase in activity under simulated solar light can be attributed to the high light absorption and generation of more electron-hole pairs by the hybrid catalyst. An increase in  $\text{gC}_3\text{N}_4$  content resulted in high photocatalytic activity up to 75 wt%  $\text{gC}_3\text{N}_4$ ; however, **TPG-100** showed lower performance than **TPG-75** (Fig. 3a).

Optimizations in PHE was performed by varying quantities of the catalyst, SED, and cocatalyst. The effect of the photocatalyst amount was evaluated with **TPG-75** under standard conditions. With 2 mg and 10 mg catalyst loading under the same optimized conditions, a maximum of  $129\,416\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  and  $38\,717\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  of HER rate was obtained, respectively (Fig. S24†). Although the rate is high for 2 mg of **TPG-75**, the cumulative production of  $\text{H}_2$  was only  $1294\ \mu\text{mol}$  compared to  $2030\ \mu\text{mol}$  with a 5 mg catalyst and  $1770\ \mu\text{mol}$  with a 10 mg catalyst. At the same time, a reduction in production was observed with 10 mg of the **TPG-75** catalyst because more catalyst concentration can hinder light absorption. After confirming the optimum catalyst amount, the quantity of SED used was optimized. Initially, 100 mg SED was used by keeping other conditions the same and tested for PHE. It was noted that by increasing the AA quantity from 100 to 200 mg, a drastic increase in the HER rate was observed. Furthermore, an increase in AA from 200 to 300 and 400 mg yielded nearly the same HER rate (Fig. S25†). Later, the quantity of PVP-Pt NPs was optimized with 5 mg of **TPG-75**, and upon increasing the loading of PVP-Pt NPs from 100 to 400  $\mu\text{L}$ , a decrease in  $\text{H}_2$  evolution was observed (Fig. S26†). The optimization experiments concluded that a combination of 5 mg of **TPG-75**, 20 mL  $\text{H}_2\text{O}$ , 200 mg of AA, and 100  $\mu\text{L}$  of PVP-Pt NPs is the optimum condition for the HER. Different sacrificial agents were screened for the PHE studies of **TPG-75**. Besides AA, triethanolamine, sodium ascorbate (SA), lactic acid, and citric acid were used for PHE. Interestingly, **TPG-75** showed a steady performance of  $26\,433\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  with SA as SED. At the same time, the hybrid catalyst did not show any  $\text{H}_2$  evolution with other SEDs (Fig. S27†). **TPG-75** exhibits a more remarkable performance than many of the reported COF catalysts in terms of PHE from deionized water (Table S2†). The enhanced HER activity has arisen from the synergistic effect of **Tp-Pa** and  $\text{gC}_3\text{N}_4$  parts in the **TPGs**.

As an extensive study, we employed our catalyst in the photocatalytic HER from water sources other than deionized water (Table S3†). First, the performance of **TPG-75** was studied in industrial wastewater containing 0.1% formic acid, propionic acid, and butyric acid. **TPG-75** could give a  $109\,432\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  HER rate under visible light irradiation and optimized conditions (Fig. 3c). This is the highest value reported for any organic photocatalyst from industrial wastewater.<sup>55</sup> Also, the performance of the same catalyst with processed seawater was  $90\,602\ \mu\text{mol g}^{-1}\text{ h}^{-1}$  which is also the best performance, higher than

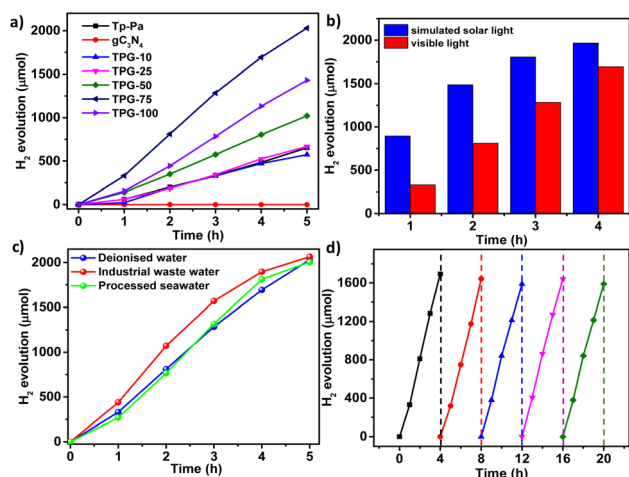


Fig. 3 (a) Comparison of the HER rate of TPG-xs with **Tp-Pa** and  $\text{gC}_3\text{N}_4$ . (b) Comparison of the HER rate of **TPG-75** with visible light and simulated solar light. (c) Comparison of the HER rate of **TPG-75** with different water sources. (d) Cycling stability of the PHE of **TPG-75** with deionized water up to 5 cycles.



that from deionized water under similar conditions of PHE. From these studies, we found that the performance of the catalyst is not degrading in wastewater and processed seawater and is stable in both cases to deliver a good HER. The reusability and cycling studies of the hybrid photocatalyst under the optimized conditions under visible light irradiation were conducted, and it was found that the photocatalyst is stable with similar performance for more than 5 cycles (Fig. 3d). The recycled sample was filtered again, recovered fully, and tested for chemical and morphological changes, if any. It is worth noting that even after repeated cycles, the catalyst remained the same without any considerable physical or chemical degradation. HR-TEM images show layered sheet-like morphology with Pt NPs of size 2–4 nm embedded in the sheet (Fig. S28†). STEM analysis and elemental mapping of the sample were also conducted which proved the incorporation of Pt NPs all over the surface and between layers of TPG-75 (Fig. S29†). In addition, Pt NPs were distributed in a segregated way all over the catalyst surface owing to the affinity of NPs to the heteroatoms of the hybrid catalyst. The FT-IR spectrum of the recycled catalyst shows no extra peak confirming the stability of the catalyst (Fig. S30†). To understand this further, the HER was performed by physically mixing **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** in the same ratio as that adopted for TPG-75 before irradiation. Interestingly, we could observe a decline in the HER compared to TPG-75 (Fig. S31†). This indicates the need to physically connect the two components to boost HER performance. The apparent quantum yield (AQY) of the best-performing sample TPG-75 was investigated at different wavelengths. For this, monochromatic wavelength filters ranging from 450, 500, and 550 nm were chosen for irradiation. Interestingly, an AQY of 10.4% was obtained for TPG-75 in deionized water at 450 nm whereas 8.34% and 4% AQY were obtained at 500 nm and 550 nm, respectively. The performance of TPG-75 was also studied under direct sunlight. In a typical experiment with the same optimized conditions, the performance of TPG-75 under direct sunlight was nearly comparable with the H<sub>2</sub> evolution using a solar simulator (Fig. S32†). However, the intensity of light irradiation controls the H<sub>2</sub> evolution in direct sunlight experiments (Fig. S32†). It is clearly understood that a similar performance of TPG-75 under a solar simulator was retained with half the light intensity under direct sunlight. The intensity of the light used from the solar simulator was 300 mW cm<sup>-2</sup> whereas the average intensity from sunlight was nearly 152 mW cm<sup>-2</sup>.

To understand the synergistic effect between pristine COF and **gC<sub>3</sub>N<sub>4</sub>** in hybrid TPG-*x*, different experiments were conducted. The steady-state emission spectrum of the hybrid photocatalyst was measured and compared with that of **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** (Fig. S33†). Accordingly, **gC<sub>3</sub>N<sub>4</sub>** shows an emission peak at around 445 nm; however, **Tp-Pa** and TPG-*x* were not emissive. In fact, the emission of the **gC<sub>3</sub>N<sub>4</sub>** counterpart in TPG-*x* was quenched due to the charge transfer (CT) between **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** and this was supported by a decrease in emission lifetime of **gC<sub>3</sub>N<sub>4</sub>** in the presence of **Tp-Pa** (Fig. S34 and Table S4†). To elucidate this assumption, we performed additional experiments by preparing control samples. Pristine **gC<sub>3</sub>N<sub>4</sub>** was ground with different ratios of **Tp-Pa** (100 : 0.01, 100 : 0.05, 100 :

1, and 100 : 5), and the corresponding emission spectra were recorded (Fig. S35†). We could observe a gradual quenching of the emission intensity of **gC<sub>3</sub>N<sub>4</sub>** upon mixing with **Tp-Pa** (Fig. S36†). The intensity of emission maxima decreased as the amount of **Tp-Pa** increased, along with a slight blue shift in emission maxima. This could point out the CT between **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>**.<sup>44,56</sup> Correspondingly, the emission lifetime of control samples was also measured (Fig. S37 and Table S5†). The band structure of **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** was confirmed by ultraviolet photoelectron spectroscopy (UPS) experiments.<sup>57</sup> From UPS studies, the energy of the valence band maximum (*E*<sub>VBM</sub>) was calculated by subtracting the width of the UPS spectrum from the excitation energy of the He I source (21.22 eV). *E*<sub>VBM</sub> of **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** was found to be −7.45 and −6.73 eV, respectively (Fig. S38†). Similarly, the energy of the conduction band minimum (*E*<sub>CBM</sub>) was calculated from the Tauc plot and *E*<sub>VBM</sub> and was found to be −5.31 and −3.63 eV, respectively.

DFT calculations were used to understand the charge distribution between **gC<sub>3</sub>N<sub>4</sub>** (001) and **Tp-Pa** (001) surfaces. First, the slab structures of **gC<sub>3</sub>N<sub>4</sub>** and **Tp-Pa** were optimized for further calculations (Fig. 4a and b). Fig. 4c and d show the electrostatic potential of **gC<sub>3</sub>N<sub>4</sub>** and **Tp-Pa** where it is observed that the work functions of **gC<sub>3</sub>N<sub>4</sub>** and **Tp-Pa** are 4.58 eV and 4.71 eV, respectively. Work function differences between the heterojunction integrant give us an idea about charge distribution and hence induce electric field formation. This in turn affects the CT process and separates photogenerated electrons and holes.<sup>58</sup> As the work function of **gC<sub>3</sub>N<sub>4</sub>** is smaller than that of **Tp-Pa**, charges will move from **gC<sub>3</sub>N<sub>4</sub>** to the COF until the

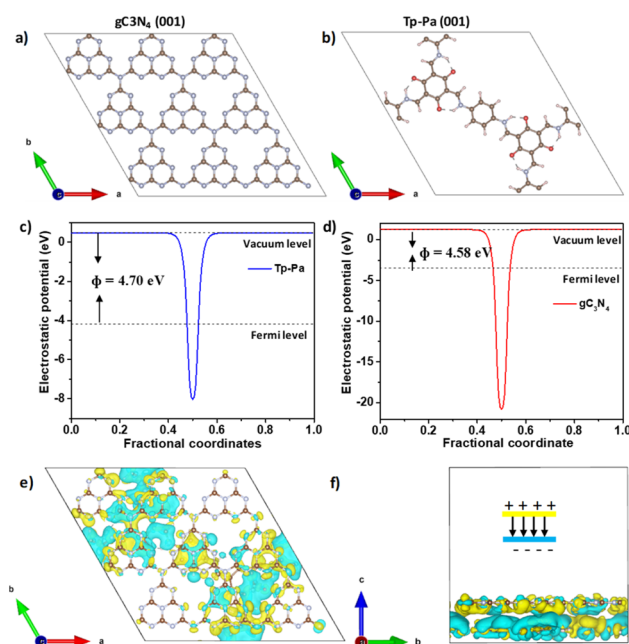


Fig. 4 Optimized slab structure of (a) **gC<sub>3</sub>N<sub>4</sub>** and (b) **Tp-Pa** COF. Theoretical electrostatic potential of (c) **Tp-Pa** (001) and (d) **gC<sub>3</sub>N<sub>4</sub>** (001). Three-dimensional calculated charge density difference of the **Tp-Pa/gC<sub>3</sub>N<sub>4</sub>** heterostructure from the (e) top view and (f) side view.



Fermi level equilibrium is reached. Also, the Fermi level position of **Tp-Pa** ( $-4.23$  eV) being lower than that of  $\text{gC}_3\text{N}_4$  ( $-3.37$  eV) will accelerate the CT from  $\text{gC}_3\text{N}_4$  to **Tp-Pa**. Furthermore, to understand the CT process, the charge density difference between **Tp-Pa** and  $\text{gC}_3\text{N}_4$  was calculated. For this calculation, a heterostructure model consisting of a  $3 \times 3$  cell of  $\text{gC}_3\text{N}_4$  (001) and one unit cell of **Tp-Pa** (001) was taken. We could see that after optimizing, there is a pronounced distortion in the surfaces resulting from the interaction between  $\text{gC}_3\text{N}_4$  and **Tp-Pa** (Fig. S39<sup>†</sup>). Also, a three-dimensional (3D) charge density differential for the **Tp-Pa/gC<sub>3</sub>N<sub>4</sub>** heterostructure was obtained to understand the CT process (Fig. 4e and f). The yellow and cyan regions show positive and negative charges on the heterostructure. It is evident from the calculation that **Tp-Pa** is mostly dominated by the cyan region (negative) and  $\text{gC}_3\text{N}_4$  is mostly dominated by the yellow region (positive) indicating a good amount of charge accumulation and reduction on **Tp-Pa** and  $\text{gC}_3\text{N}_4$ , respectively. When in contact, the charge accumulation and reduction occur and ultimately cause a band edge bending phenomenon on  $\text{gC}_3\text{N}_4$  and **Tp-Pa**, respectively. The band bending phenomenon causes the band edge of **Tp-Pa** to move downwards whereas the band edge of  $\text{gC}_3\text{N}_4$  moves upwards and the PHE of **TPG-xs** can be explained based on these conclusions (Fig. 5). Upon irradiating with light, the electrons of  $\text{gC}_3\text{N}_4$  and **Tp-Pa** will get excited to higher energy levels. Due to the band bending up/downward phenomenon, the excited electrons of **Tp-Pa** will recombine with the holes of  $\text{gC}_3\text{N}_4$ , hence decreasing the charge recombination process within  $\text{gC}_3\text{N}_4$ . Furthermore, the excited state electrons of  $\text{gC}_3\text{N}_4$  will be transferred to Pt NPs. At the same time, the holes generated in **Tp-Pa** will be carried forward by AA and oxidized to dehydroascorbic acid. The above observation was further confirmed by PHE studies and emission studies of control samples (Fig. S36<sup>†</sup>). It must be remembered that  $\text{gC}_3\text{N}_4$  is not performing well with AA suggesting that whatever holes were generated in  $\text{gC}_3\text{N}_4$  could not be transferred to AA and oxidized. Also, from emission, it was evident that due to CT between  $\text{gC}_3\text{N}_4$  and **Tp-Pa**, the emission of  $\text{gC}_3\text{N}_4$  was quenched by adding **Tp-Pa**.

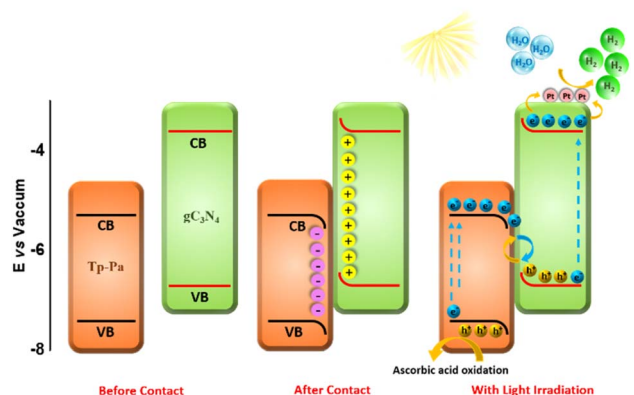


Fig. 5 Band structure of **Tp-Pa** and  $\text{gC}_3\text{N}_4$  with a plausible PHE mechanism upon irradiating with light.

We prepared **TPG-75** on a 20 g scale using a planetary mixer and the detailed procedure is provided in the experimental section. After extensive washing and purification, **TPG-75-20G** was characterized by  $^{13}\text{C}$  NMR, FT-IR, XPS, BET, HR-TEM, PXRD (Fig. S40<sup>†</sup>), FE-SEM, and STEM elemental mapping (Fig. S41<sup>†</sup>). All the experimental results are in line with the small-scale synthesis (2 g) and confirm the suitability of the synthesis protocol for the large-scale production of **TPGs**. PHE of **TPG-75-20G** exhibited a comparable performance of  $84\,383\,\mu\text{mol g}^{-1}\text{ h}^{-1}$  with deionized water and no significant reduction in the  $\text{H}_2$  evolution rate upon scale-up synthesis (Fig. 6a and b, and Video S1<sup>†</sup>). Similarly, comparable performance to that of **TPG-75** was noticed with processed seawater ( $90\,371\,\mu\text{mol g}^{-1}\text{ h}^{-1}$ ) and industrial wastewater ( $94\,873\,\mu\text{mol g}^{-1}\text{ h}^{-1}$ ) (Fig. 6b). To further extend the scope of using the catalyst for the production of  $\text{H}_2$  from seawater, we prepared simulated seawater using a standard protocol<sup>59</sup> and examined it for PHE studies. Using optimized PHE conditions, the performance of **TPG-75-20G** from simulated seawater was found to be  $109\,125\,\mu\text{mol g}^{-1}\text{ h}^{-1}$  which is one of the highest reported values among different photocatalysts studied to date (Fig. 6a). The exceptional increment in the HER rate from simulated seawater can be attributed to the *in situ* polarization of the framework arising from the adsorption of metal salts. The polarization effect plays a crucial role in amplifying the dielectric constant of the organic semiconductor. This, in turn, reduces the exciton dissociation energy, enhancing the charge separation and transfer. Consequently, it facilitates the promotion of  $\text{H}_2$  production from seawater.<sup>49</sup> We compared these results with reported literature on the photocatalytic HER from seawater and found that our sample is one of the best-performing catalysts among organic, inorganic, and hybrid materials (Fig. 6c and Table S6<sup>†</sup>). The cycling stability of **TPG-75-20G** was also investigated, revealing stability for more than 5 cycles with minimal degradation observed in hydrogen evolution using deionized water (Fig. S42<sup>†</sup>).

To further support the results, we have obtained photocurrent responses and conducted EIS studies of **TPG-75-20G** in different water sources. From photocurrent responses, we could

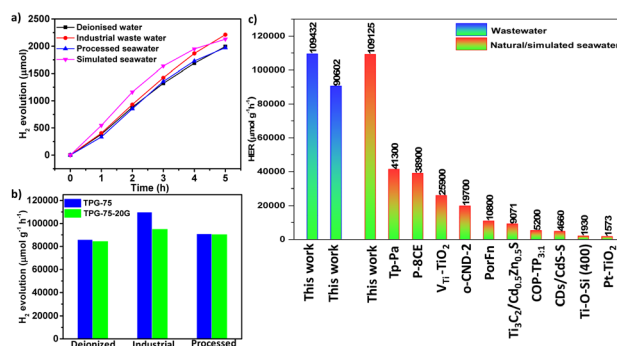


Fig. 6 (a) HER performance of **TPG-75-20G** with different water sources and (b) comparison of HER performance of **TPG-75** and **TPG-75-20G** with different water sources. (c) Comparison of the HER rate of **TPG-75** with that of different hybrid materials reported so far from wastewater and simulated or natural seawater.



see that the light responses of **TPG-75-20G** were more prominent in industrial wastewater and simulated seawater compared to deionized water (Fig. S43†) proving the higher  $H_2$  evolution from industrial wastewater and simulated seawater. The EIS studies also revealed small charge transfer resistance for **TPG-75-20G** with different water sources (Fig. S43†). In short, all the supporting experiments point to the merit of our work to achieve scalable synthesis and remarkable  $H_2$  production from seawater and industrial wastewater.

## Conclusions

We present, for the first time, the large-scale synthesis of an organic photocatalyst with outstanding HER efficiency. We successfully synthesized a hybrid photocatalyst based on **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** through a scalable mechanochemical method and comprehensively characterized it using various experimental techniques. The hybrid photocatalyst exhibited high crystallinity and surface area. Our comprehensive research, incorporating electrochemical methods and theoretical calculation, suggests that the presence of both **Tp-Pa** and **gC<sub>3</sub>N<sub>4</sub>** components results in synergistic effects. Consequently, the semi-conducting properties of the hybrid catalyst can be finely tuned, making it a promising photocatalyst for  $H_2$  evolution from water. The improved photocatalytic HER rate can be attributed to the enhanced charge dissociation and exciton formation within the hybrid catalyst. Notably, the hybrid catalyst achieves a remarkable photocatalytic HER rate of 85 617 and 179 064  $\mu\text{mol g}^{-1} \text{h}^{-1}$  under visible light and simulated solar light, respectively. It also performs exceptionally well with different water sources, demonstrating rates of 109 432  $\mu\text{mol g}^{-1} \text{h}^{-1}$  for industrial wastewater and 90 602  $\mu\text{mol g}^{-1} \text{h}^{-1}$  for processed seawater. This represents the highest reported rate for a photocatalyst that is effective in deionized water, wastewater, and seawater.

Furthermore, we have explored the bulk synthesis of these photocatalysts, a critical step toward potential industrial applications. A facile mechanochemical synthesis procedure using a planetary mixer enabled us to successfully synthesize 20 g of photocatalyst in a single batch while retaining crystallinity, surface area, and nearly the same  $H_2$  evolution. Remarkable HER rates of 84 383, 90 371, 94 873, and 109 125  $\mu\text{mol g}^{-1} \text{h}^{-1}$  were obtained for deionized water, processed seawater, industrial wastewater, and simulated seawater, respectively. Our work contributes valuable insights into the development of organic semiconductors for industrial applications, emphasizing scalability and mass production targeting potential commercialization.

## Data availability

The datasets supporting this article have been uploaded as part of the ESI.†

## Author contributions

S. S. B. conceived the research idea. K. A. synthesized and characterized COFs and conducted all photocatalytic HER

experiments. B. T. M. designed and performed all the electrochemistry experiments. G. D. and S. K. carried out the DFT calculation and provided the corresponding write-up. The results were regularly discussed and all the contributing authors commented and provided significant suggestions. K. A. and S. S. B. wrote the manuscript with contributions from all authors.

## Conflicts of interest

There are no conflicts to declare.

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