



Cite this: *Chem. Sci.*, 2024, **15**, 10084

All publication charges for this article have been paid for by the Royal Society of Chemistry

Dual-strategy engineered nickel phosphide for achieving efficient hydrazine-assisted hydrogen production in seawater†

Rui-Qing Li, ^{*a} Songyun Guo, ^a Xiaojun Wang, ^a Xiaoyu Wan, ^a Shuixiang Xie, ^a Yu Liu, ^a Changming Wang, ^a Guangyu Zhang, ^a Jun Cao, ^{*b} Jiamu Dai, ^a Mingzheng Ge ^a and Wei Zhang ^{*a}

Electrocatalytic hydrogen production in seawater to alleviate freshwater shortage pressures is promising, but is hindered by the sluggish oxygen evolution reaction and detrimental chloride electrochemistry. Herein, a dual strategy approach of Fe-doping and CeO₂-decoration in nickel phosphide (Fe-Ni₂P/CeO₂) is rationally designed to achieve superior bifunctional catalytic performance for the hydrogen evolution reaction (HER) and hydrazine oxidation reaction (HzOR) in seawater. Notably, the two-electrode Fe-Ni₂P/CeO₂-based hybrid seawater electrolyzer realizes energy-efficient and chlorine-free hydrogen production with ultralow cell voltages of 0.051 and 0.597 V at 10 and 400 mA cm⁻², which are significantly lower than those needed in the hydrazine-free seawater electrolyzer. Density functional theory calculations manifest that the combination of Fe doping and heterointerface construction between Fe-Ni₂P and CeO₂ can adjust the electronic structure of the Ni₂P and optimize the water dissociation barrier and hydrogen adsorption free energy, leading to improvement of the intrinsic catalytic performance. This route affords a feasible solution for future large-scale hydrogen generation using abundant ocean water.

Received 19th February 2024
Accepted 26th April 2024

DOI: 10.1039/d4sc01160k
rsc.li/chemical-science

Introduction

Hydrogen, with its high energy-density and pollution-free emissions, is a superior energy source for the realization of carbon neutrality. Water electrolysis driven by renewable electricity is a prospective scheme to prepare hydrogen.^{1,2} In general, water electrolyzers employ large quantities of freshwater as the electrolyte, restricting their grid-scale deployment.³ In this context, substituting freshwater with the abundant seawater of our planet would provide an unlimited hydrogen source and alleviate freshwater shortage pressures. Seawater electrolysis involves two half reactions, namely, the hydrogen evolution reaction (HER) and the oxygen evolution reaction (OER), and the OER process limits the overall electrolytic efficiency because of its intrinsically sluggish kinetics, which lead to high cell voltages.^{4,5} Unfortunately, the chloride ions present inevitably undergo the chlorine evolution reaction (CER), which competes with the OER at the anode, bringing about the generation of corrosive chlorine species when the overpotential

exceeds 480 mV, catalyst degradation and environmental hazards in alkaline seawater.^{6,7} Consequently, developing chlorine-free and energy-saving seawater electrolysis technology for sustainable hydrogen production remains a formidable challenge.⁸ At present, two tactics have been explored to solve the above bottlenecks. Firstly, replacing the OER with thermodynamically favorable oxidation reactions of small molecules including urea, formate, hydrazine and furfural can markedly decrease electrolysis voltages and avoid the occurrence of the CER.^{9–12} Among these, the hydrazine oxidation reaction (HzOR, N₂H₄ + 4OH[–] → N₂ + 4H₂O + 4e[–]) possesses a low theoretical potential (–0.33 V vs. RHE) compared to the OER and CER and has environmentally friendly products, which endows it with promising application prospects.^{13,14} To the best of our knowledge, hydrazine is extensively considered to be an industrially toxic substance, and its disposal has adverse consequences on humans and the environment. The electrocatalytic HzOR process not only eliminates hydrazine from industrial sewage without adding oxidants or performing complicated separation treatments, but also contributes to the generation of chlorine-free and energy-efficiency hydrogen when coupled with the HER in seawater.^{15,16}

Exploiting active and steady catalysts is the second crucial solution to improve catalytic properties and overcome chlorine electrochemistry during seawater electrolysis.^{17–19} Currently, noble-metal materials exhibit remarkable intrinsic activities

^aSchool of Textile and Clothing, Nantong University, Nantong 226019, PR China.
E-mail: liuruqing@ntu.edu.cn; zhangwei@ntu.edu.cn

^bDepartment of Physics, Zhejiang Sci-Tech University, Hangzhou 310018, PR China.
E-mail: caojun2021@zstu.edu.cn

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4sc01160k>



due to their special electronic structure and proper adsorption energy with reactants and intermediates.^{20,21} Nevertheless, their scarcity, high price and general stability impede their large-scale industrialization. It is thus meaningful to explore abundant and cheap catalysts such as oxides, phosphides, nitrides and sulfides.^{22–25} Phosphides, which exhibit unique structures, controllable composition and eminent electrical conductivity, represent a class of effective and promising substitutes.^{26,27} To the best of our knowledge, many phosphides display ordinary catalytic performances that cannot satisfy commercial demands, especially at industrial-grade current densities, due to their single and underexposed catalytic sites and inappropriate binding strength between intermediates and catalytic sites.²⁸ Some pioneering works have committed to optimizing catalytic properties *via* various strategies, including heteroatom doping, modulation of the heterostructure and local coordination environment, construction of hierarchical structures, *etc.*^{29–31} Heteroatom doping can effectively regulate energy levels and improve catalytic activities without altering the initial crystal structure.³² For instance, introducing W and P into Co₃N manipulates the catalytic kinetics and reduces the hydrogen production voltage.³³ Although significant progress has been made, most developed catalysts are not capable of possessing satisfactory bifunctional HER and HzOR activities on account of incompatible catalytic sites. In addition, the catalytic performances can be enhanced by adding valence-changing cocatalysts to redistribute the electrons of the catalysts.^{34,35} Cerium oxides, with their high oxygen storage capacity and remarkable ability to gain and lose electrons, are known as effective promoters to advance catalytic activities.³⁶ Therefore, it is promising to integrate heteroatom doping and cerium oxide coupling to adjust the electronic structure and obtain numerous catalytic sites, giving rise to remarkable the HER and HzOR performances of phosphides to realize energy-efficient hydrogen generation.

Herein, we developed self-supported Ni₂P nanosheet arrays with Fe doping and CeO₂ decoration on a Ni foam substrate (Fe-Ni₂P/CeO₂); these arrays achieve energy-efficient and chlorine-free hydrogen production in a hydrazine-assisted seawater electrolyzer. The Fe-Ni₂P/CeO₂ displays splendid bifunctional catalytic activities for the HER and HzOR due to the electronic structure modification *via* metal doping and heterostructure construction, as well as uniform nanosheet arrays. The two-electrode hybrid seawater system with Fe-Ni₂P/CeO₂ requires low operating voltages of 0.051 and 0.597 V to attain 10 and 400 mA cm⁻², which are significantly lower than the corresponding voltages of conventional seawater electrolysis, indicating its bright prospects for producing hydrogen.

Results and discussion

Synthesis and characterization of Fe-Ni₂P/CeO₂

The self-supported Fe-Ni₂P/CeO₂ nanosheet arrays on the NF substrate were obtained by hydrothermal and phosphorization reactions, as shown in Fig. 1a. Firstly, the NiFeCe hydroxide precursor was uniformly grown on the surface of NF; the resulting X-ray diffraction (XRD) pattern (Fig. S1†) exhibits

several characteristic peaks of the layered double hydroxide (NiFeCe LDH).³⁷

The morphology and structure of the NiFeCe LDH precursor were analyzed using scanning electron microscopy (SEM), and it was found to be composed of a large number of interconnected nanosheets (Fig. 1b and c). After the phosphorization treatment, we obtained Fe-Ni₂P/CeO₂ without other impurities as verified *via* XRD analysis (Fig. 1d). The characteristic peaks at 28.5° and 56.3° and at 40.7°, 47.4°, 54.3° were indexed to the (111) and (311) planes of CeO₂ (JCPDS no. 34-0394) and the (111), (210), and (300) crystalline planes of Ni₂P (JCPDS no. 74-1385), which correspond to Ni₂P and Fe-Ni₂P samples, suggesting the formation of Fe-Ni₂P/CeO₂. Moreover, it can be observed that the Fe-Ni₂P/CeO₂ retains the nanosheet-like morphology of NiFeCe LDH distributed throughout the NF skeleton (Fig. 1e and S2†), which is similar to that of Ni₂P (Fig. S3a†) and Fe-Ni₂P (Fig. S3b†), which is beneficial for exposing abundant active sites and providing fast electrolyte transport. For detailed analysis of the microstructure, transmission electron microscopy (TEM) measurement was carried out, and the nanosheet-like structure of Fe-Ni₂P/CeO₂ can also be seen (Fig. 1f), consistent with SEM results. In Fig. 1g, the high-resolution TEM (HRTEM) image exhibits two distinct lattice fringes with interplanar spacings of 0.206 and 0.313 nm, which were assigned to the (201) and (111) planes of Fe-Ni₂P and CeO₂, revealing the successful construction of the heterostructure. The elemental composition and distribution of Fe-Ni₂P/CeO₂ were studied. In Fig. 1h–m, the SEM and corresponding elemental mapping images indicate that the Fe-Ni₂P/CeO₂ is comprised of the elements Ni, Fe, P, Ce and O, which are homogeneously dispersed over the entire nanosheets.

The chemical status and electronic effects of Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂ were identified using X-ray photoelectron spectroscopy (XPS). As shown in Fig. 2a, the Ni 2p spectrum of Ni₂P has two pairs of typical spin-orbit peaks centered at 854.3 and 855.5 eV and at 872.4 and 873.4 eV, corresponding to Ni 2p_{3/2} and Ni 2p_{1/2}.³⁸ The peaks located at 861.8 and 880.0 eV are assigned as satellite peaks. Noteworthily, the Ni 2p spectrum of Fe-Ni₂P (Fig. 2a) shows a positive shift of 0.4 eV compared with Ni₂P, indicating that the introduction of Fe can modify the electronic structure of Ni₂P. After coupling CeO₂, the Ni 2p peaks of Fe-Ni₂P/CeO₂ shift to a more positive binding energy of 0.3 eV, suggesting further optimization of the electronic structure of Fe-Ni₂P and strong electronic interaction between Fe-Ni₂P and CeO₂, which are beneficial to boost catalytic performances.³⁹ Similar to the Ni 2p spectrum, the Fe 2p spectrum of Fe-Ni₂P (Fig. 2b) can be fitted to two Fe doublets, with the peaks at 709.9 and 712.9 eV belonging to Fe 2p_{3/2} and the peaks at 722.8 and 725.7 eV corresponding to Fe 2p_{1/2}. The signals of satellite peaks at 717.7 and 732.3 eV can be observed. After decoration with CeO₂, the Fe 2p peaks of Fe-Ni₂P/CeO₂ display a redshift of 0.4 eV compared to those of Fe-Ni₂P, proving the existence of a coupling effect and electronic transfer between Fe-Ni₂P and CeO₂.⁴⁰ In Fig. 2c, the peaks of P 2p at 128.9 and 129.7 eV are assigned to spin-splitting P 2p_{3/2} and P 2p_{1/2}, indicating phosphide formation. The presence of P–O bonds (133.8 eV) is observed, probably because of the surface oxidation



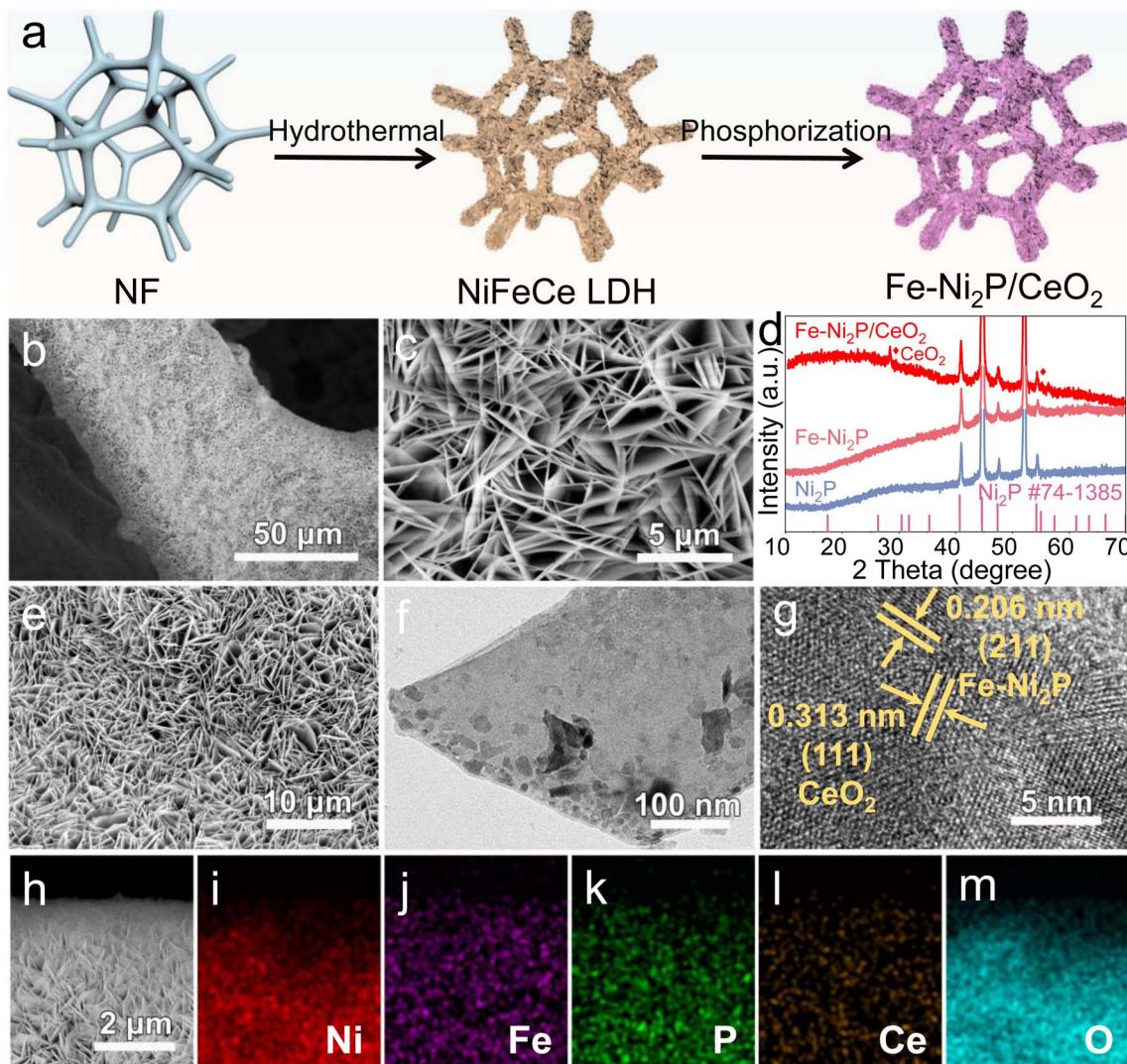


Fig. 1 (a) Schematic illustration of the preparation process for Fe-Ni₂P/CeO₂. (b and c) SEM images of NiFeCe LDH. (d) XRD patterns of Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂. (e) SEM, (f) TEM, (g) HRTEM, (h–m) SEM and corresponding element mapping images for Fe-Ni₂P/CeO₂.

of phosphide exposed to air.⁴¹ In the Ce 3d spectrum (Fig. 2d), Ce 3d_{5/2} and 3d_{3/2} peaks can be observed.

Electrochemical performance of Fe-Ni₂P/CeO₂

To explore the effect of Fe doping and CeO₂ coupling on catalytic performance, the catalytic HER, OER and HzOR performances of Fe-Ni₂P/CeO₂ were evaluated in alkaline seawater, and comparative experiments for Ni₂P, Fe-Ni₂P and commercial Pt/C were also conducted. Fig. 3a displays the HER polarization curves of Ni₂P, Fe-Ni₂P, Fe-Ni₂P/CeO₂ and Pt/C, and Pt/C possesses exceptional HER performance. Ni₂P has general HER activity requiring a high overpotential of 157 mV to deliver 10 mA cm⁻². After the introduction of Fe, the overpotential of Fe-Ni₂P is reduced to 101 mV to attain the same current compared with Ni₂P, indicating the enhanced HER performance of Fe-Ni₂P. Inspiringly, the Fe-Ni₂P/CeO₂ displays an eminent activity with a low overpotential of 67 mV at 10 mA cm⁻² (Fig. 3a and f), surpassing those of Ni₂P, Fe-Ni₂P and most exploited transition

metal materials (Fig. 3g and Table S1†). Moreover, the Fe-Ni₂P/CeO₂ can output large current densities of 200 and 400 mA cm⁻² at small overpotentials of 197 and 249 mV, which are lower than those of Ni₂P (315 and 379 mV) and Fe-Ni₂P (234 and 301 mV). These results indicate that cation Fe doping and CeO₂ coupling play crucial roles in the promotion of HER performances. The corresponding HER kinetics were assessed using the Tafel slope derived from the polarization curves.⁴² The Tafel slope value of Fe-Ni₂P/CeO₂ (Fig. 3b and f) was calculated to be 53 mV dec⁻¹, which was smaller than those of Ni₂P (108 mV dec⁻¹) and Fe-Ni₂P (64 mV dec⁻¹), affirming that the introduction of Fe and CeO₂ can accelerate the HER kinetics and that the HER process of Fe-Ni₂P/CeO₂ follows a Volmer–Heyrovsky process. To examine the origin of the high catalytic activities for Fe-Ni₂P/CeO₂, the double layer capacitance (C_{dl}) was computed from the related cyclic voltammetry (CV) curves (Fig. S4†) at various scan rates, which reflect the electrochemical surface area (ECSA) because of the proportional relationship.⁴³ In Fig. 3c and f, Fe-

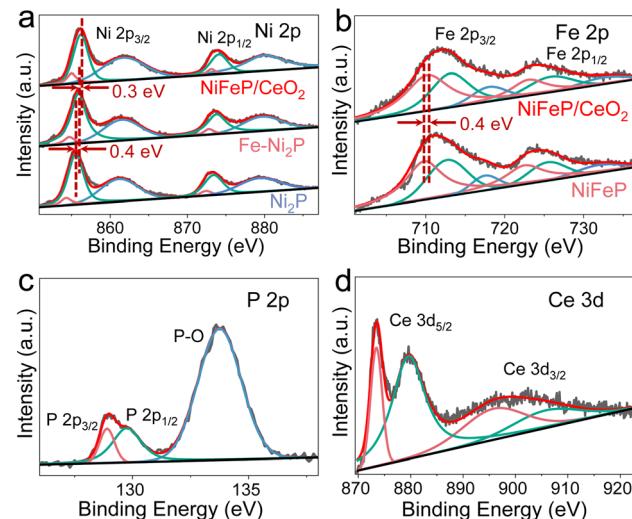


Fig. 2 (a) Ni 2p spectra of Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂. (b) Fe 2p spectra of Fe-Ni₂P and Fe-Ni₂P/CeO₂. (c) P 2p and (d) Ce 3d spectra of Fe-Ni₂P/CeO₂.

Ni₂P/CeO₂ displays a larger C_{dl} value (31.4 mF cm⁻²) than that of Ni₂P (11.5 cm⁻²) and Fe-Ni₂P (23.5 cm⁻²), in agreement with the outstanding activity of Fe-Ni₂P/CeO₂. The computed ECSA value of Fe-Ni₂P/CeO₂ (785 cm²) is larger than that of Ni₂P (287.5 cm²) and Fe-Ni₂P (587.5 cm²), indicating the exposure of plentiful catalytic sites. In Fig. 3d, the ECSA-normalized polarization curves show that Fe-Ni₂P/CeO₂ possesses high catalytic performance, verifying that the introduction of Fe and CeO₂ can boost the intrinsic activities of active sites.⁴⁴ The charge transfer process of Fe-Ni₂P/CeO₂ was probed using electrochemical impedance spectroscopy (EIS). As shown in Fig. 3e and f, the Nyquist diagram of Fe-Ni₂P/CeO₂ has the smallest semicircle radius compared to Ni₂P and Fe-Ni₂P, and the Fe-Ni₂P/CeO₂ possesses the lowest charge-transfer resistance (R_{ct} , 3.0 Ω) and fastest electron-transfer capability at the electrode and electrolyte interface in the catalytic reaction compared with Ni₂P (4.2 Ω) and Fe-Ni₂P (3.7 Ω). We also assessed the HER activity of Fe-Ni₂P/CeO₂ in neutral seawater. Fe-Ni₂P/CeO₂ was found to possess high faradaic efficiency by comparing the amount of gas actually produced with the theoretical gas amount (Fig. S5†), indicating that the catalytic current is derived entirely from the intended catalytic reaction. In Fig. S6a,† it is shown

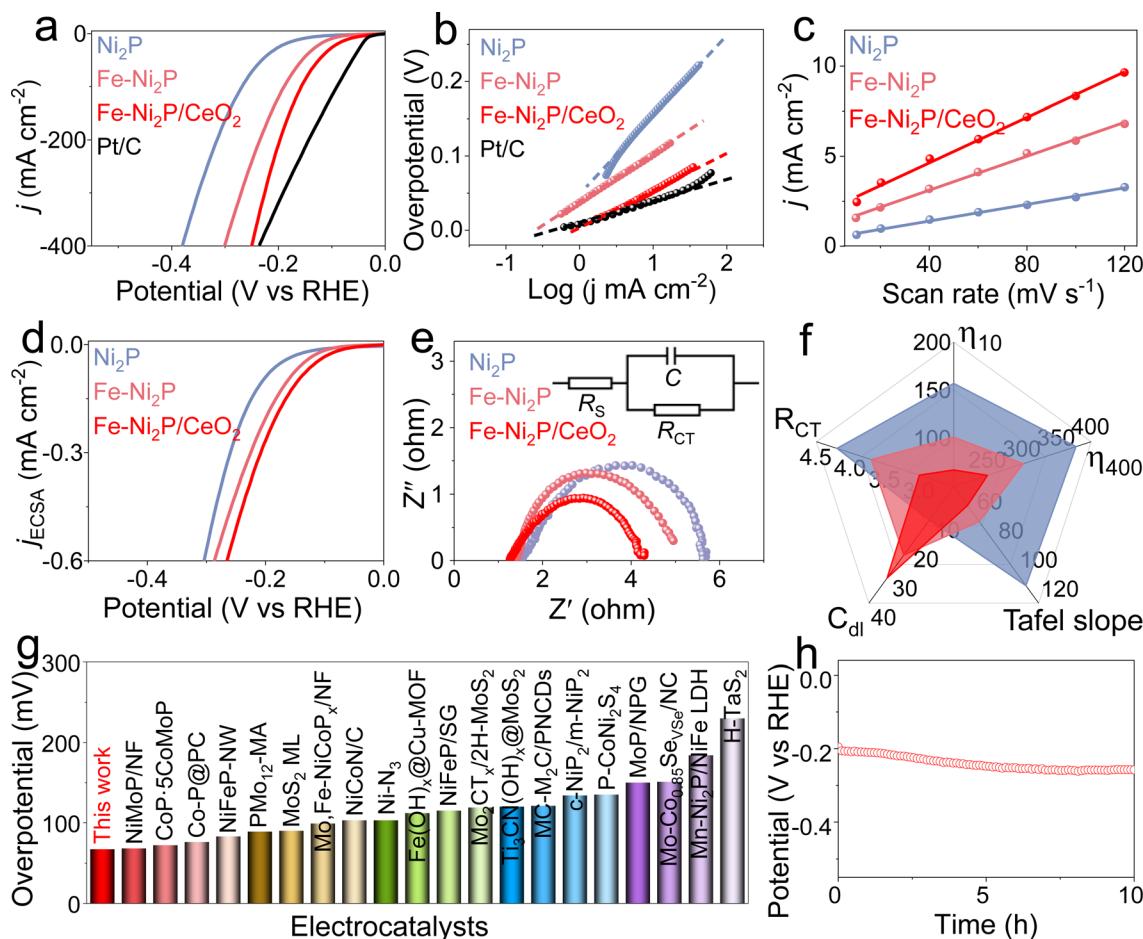


Fig. 3 (a) Polarization curves and (b) Tafel slopes for Ni₂P, Fe-Ni₂P, Fe-Ni₂P/CeO₂ and Pt/C. (c) C_{dl} values, (d) ECSA-normalized polarization curves, (e) EIS and (f) HER performance radar chart for Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂. (g) Comparison of HER performances of Fe-Ni₂P/CeO₂ with the developed catalysts. (h) Stability test for Fe-Ni₂P/CeO₂.

that $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ exhibits good catalytic performance with a small overpotential of 161 mV at 10 mA cm^{-2} in 1.0 M PBS buffered seawater solution. The catalytic durability of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ is a vital criterion to evaluate its commercialization in seawater. As shown in Fig. 3h and S6b,[†] the v - t curves of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ indicate that the needed potential does not increase significantly at a constant high current density of 200 mA cm^{-2} during continuous measurement for 10 h, indicating no evident catalytic activity loss and outstanding stability in alkaline and neutral seawater. After the test, the phase, morphology and valence states were analyzed. The XRD (Fig. S7[†]) and SEM images (Fig. S8[†]) show that the phase and nanosheet arrays of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ are well maintained, and the corresponding elements Ni, Fe, P, Ce and O are evenly distributed on the surface of the nanosheets (Fig. S9[†]). As shown in Fig. S10,[†] the binding energies of Ni 2p, Fe 2p and Ce 3d exhibit no obvious change, and the peak intensity of the P 2p spectrum diminishes, probably because P was partially dissolved in the electrolyte during long-term operation.⁴⁵

Subsequently, the OER and HzOR activities of Ni_2P , $\text{Fe-Ni}_2\text{P}$, $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ and RuO_2 were investigated in seawater media with

and without N_2H_4 . For the OER, Ni_2P possesses the poorest activity (Fig. 4a). After introducing Fe and CeO_2 , the $\text{Fe-Ni}_2\text{P}$ and $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ display sequentially improved OER performances with low potentials of 1.458 and 1.442 V at 10 mA cm^{-2} , respectively. Even so, the required potentials for $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ at the anode are still high, leading to large voltages for water electrolysis when coupled with cathodic HER. To settle this situation, the thermodynamically favorable HzOR can be employed to substitute OER. In Fig. 4b, the HzOR activities of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ were tested in 1 M KOH seawater with various N_2H_4 concentrations. Compared with the OER, the HzOR curves of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ display a sharply increased oxidation current as the N_2H_4 concentration is raised from 0 M to 0.1 M. The HzOR performance continues to increase rapidly as the N_2H_4 concentration is increased to 0.5 M; subsequently, the HzOR performance remains basically unchanged with further increasing the N_2H_4 concentration from 0.5 to 0.7 M. Thus, the HzOR activities of Ni_2P , $\text{Fe-Ni}_2\text{P}$ and RuO_2 were tested in 1 M KOH seawater with 0.5 M N_2H_4 . As shown in Fig. 4c and d, the working potential of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ is as low as -117 mV at 10 mA cm^{-2} , which is lower than those of Ni_2P (-67 mV), $\text{Fe-Ni}_2\text{P}$ (-104 mV), and RuO_2 (41 mV), revealing the

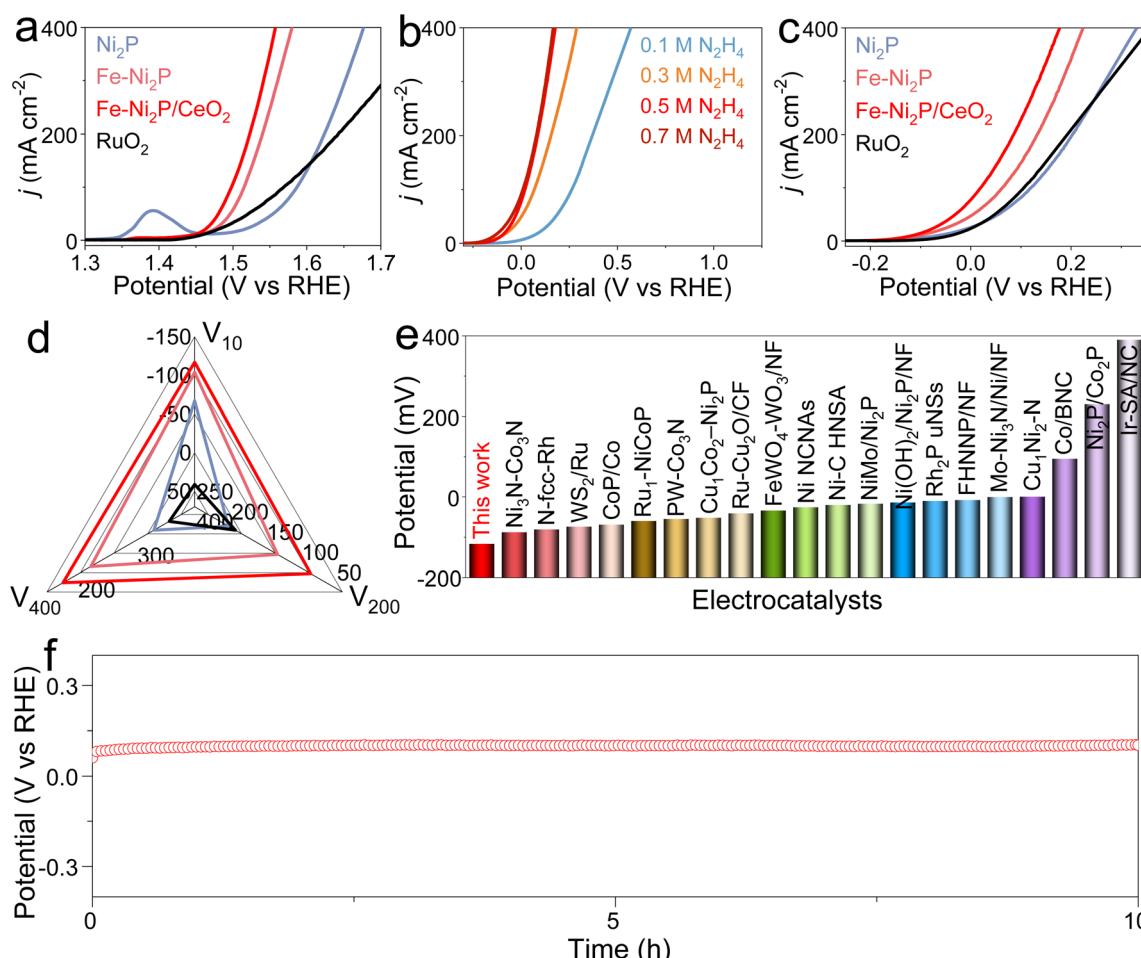


Fig. 4 (a) OER and (c) HzOR polarization curves for Ni_2P , $\text{Fe-Ni}_2\text{P}$, $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ and RuO_2 . (b) HzOR curves for $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ at various N_2H_4 concentrations. (d) Potential comparison of the HzOR at 10, 200 and 400 mA cm^{-2} for Ni_2P , $\text{Fe-Ni}_2\text{P}$, $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ and RuO_2 . (e) Comparison of the HzOR performances of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ with recently developed catalysts. (f) Stability test for $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$.

importance of Fe doping and CeO_2 decoration. Impressively, $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ requires lower potentials of 85 and 177 mV to realize large current densities of 200 and 400 mA cm^{-2} compared with Ni_2P (202 and 331 mV), $\text{Fe-Ni}_2\text{P}$ (132 and 224 mV), and RuO_2 (192 and 357 mV). The HzOR performance of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ is superior to many reported values (Fig. 4e and Table S2[†]). It is worth noting that the required HzOR potential of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ shows no obvious variation based on the $v-t$ curve (Fig. 4f), indicating its remarkable stability and corrosion resistance. After determination of the HzOR stability, the XRD pattern (Fig. S11[†]) and SEM images (Fig. S12[†]) reveal that the phase and uniform nanosheet morphology of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ are well preserved and the elements Ni, Fe, P, Ce and O were distributed over the nanosheets of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ in the elemental mapping measurement (Fig. S13[†]). The XPS spectra (Fig. S14[†]) indicate that the binding energies of the elements Ni, Fe, P, Ce and O are basically unchanged except for the reduced peak intensity of the P 2p spectrum due to the partial dissolution of phosphide at a high oxidation potential. These results suggest that $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ has outstanding corrosion resistance and structural stability, and can be regarded as an active HzOR catalyst in seawater.

To confirm the application prospects of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ with remarkable HER, OER, and HzOR properties, two-electrode OWS and OHZS electrolyzers engaging $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ as both cathode and anode were assembled in 1.0 M KOH seawater medium with and without 0.5 M N_2H_4 (Fig. 5a). As shown in Fig. 5b and c, the overall water splitting (OWS) system displays good catalytic activities with low cell voltages of 1.582, 1.863 and 1.949 V to reach 10, 200 and 400 mA cm^{-2} , which are close to or even smaller than those of many developed catalysts (Fig. S15 and Table S3[†]). It should be pointed out that the competitive CER may occur and produce corrosive chlorine species at high current densities of 200 and 400 mA cm^{-2} , resulting in catalyst corrosion and environmental hazards. After adding N_2H_4 , the assembled OHZS electrolyzer coupling HzOR with HER not only shows a significant activity enhancement, but also realizes chlorine-free H_2 generation at low voltages (Fig. 5b and c). In detail, this electrolyzer requires smaller voltages of 0.051, 0.438 and 0.597 V to deliver the same current densities compared with those needed in the OWS system and many published works (Fig. 5e and Table S4[†]). Therefore, the hybrid seawater electrolysis by adding the N_2H_4 to the electrolyte can greatly reduce hydrogen generation voltages and boost energy efficiency. Meanwhile, the $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ -based OHZS system possesses excellent long-term stability at a near-industrial current density of 400 mA cm^{-2} after continuous electrolysis for 10 h (Fig. 5d). After the durability test, the nanosheet-like morphologies of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ are well retained (Fig. S16 and S17[†]), indicating its remarkable structural robustness and corrosion resistance. These results prove that the $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ can serve as a promising catalyst to achieve sustainable hydrogen production with high energy-efficiency and chlorine-free products in seawater.

Theoretical analysis

To investigate the possible mechanism for the superior catalytic performance of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ in depth, density functional

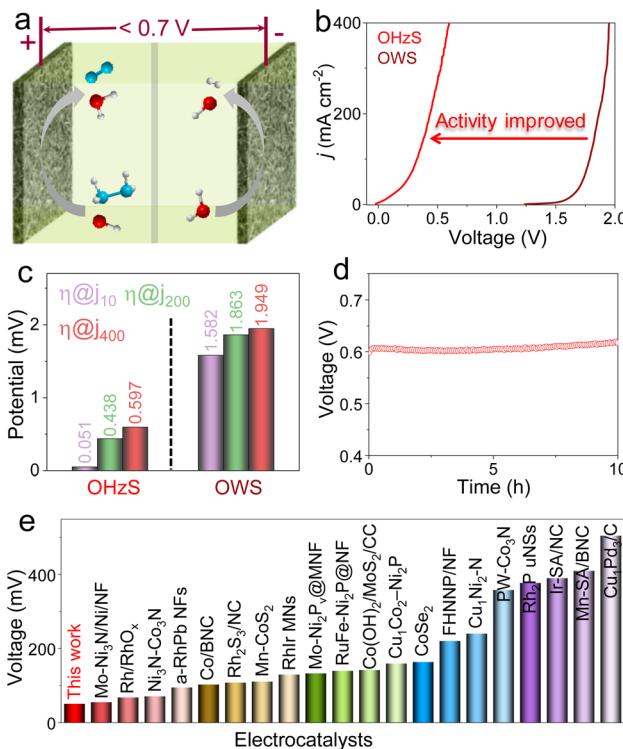


Fig. 5 (a) Schematic illustration of two-electrode OHZS electrolyzer constructed by $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$. (b) OWS and OHZS polarization curves for $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ and (c) corresponding voltage comparison at 10, 200 and 400 mA cm^{-2} . (d) Stability test for $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ at a constant current density of 400 mA cm^{-2} in 1.0 M KOH with 0.5 M N_2H_4 . (e) Comparison of OHZS performances of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ with those of other developed catalysts.

theory (DFT) calculations were carried out. The Ni_2P , $\text{Fe-Ni}_2\text{P}$ and $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ models are illustrated in Fig. 6a. The H_2O dissociation step is known to require a high energy barrier and to dominate the qualitative trend of HER kinetics under alkaline electrolyte.⁴⁶ The optimized structures for H_2O dissociation on the surface of Ni_2P , $\text{Fe-Ni}_2\text{P}$ and $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ are displayed in Fig. S18–S20.[†] Fig. 6b exhibits the calculated H_2O dissociation energies on the catalyst surfaces, and Ni_2P has a large energy barrier of 0.37 eV for H_2O dissociation and an adsorbed hydrogen (H^*) generation barrier of 0.78 eV from broken H_2O molecular bonds. After Fe doping, the corresponding energy barriers of $\text{Fe-Ni}_2\text{P}$ decreased to 0.20 and -0.11 eV, respectively. The $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ possesses the smallest activation barrier of -0.11 eV and H^* formation barrier of -0.25 eV after the coupling of CeO_2 , implying its rapid and thermodynamically favorable H_2O dissociation kinetics to generate H^* and support efficient HER. Furthermore, the free energy of H^* (ΔG_{H^*}) is a pivotal descriptor to assess HER performance, and the ideal ΔG_{H^*} should be close to thermoneutrality.⁴⁷ The optimized H^* models on the catalysts are presented in Fig. S14–S16[†] and the ΔG_{H^*} value of $\text{Fe-Ni}_2\text{P}/\text{CeO}_2$ (Fig. 6c) is calculated to 0.17 eV, which is closer to thermoneutrality compared to that of Ni_2P (0.52 eV) and $\text{Fe-Ni}_2\text{P}$ (0.48 eV), indicating accelerated hydrogen production capacity from the H^* . The reduced energy barriers



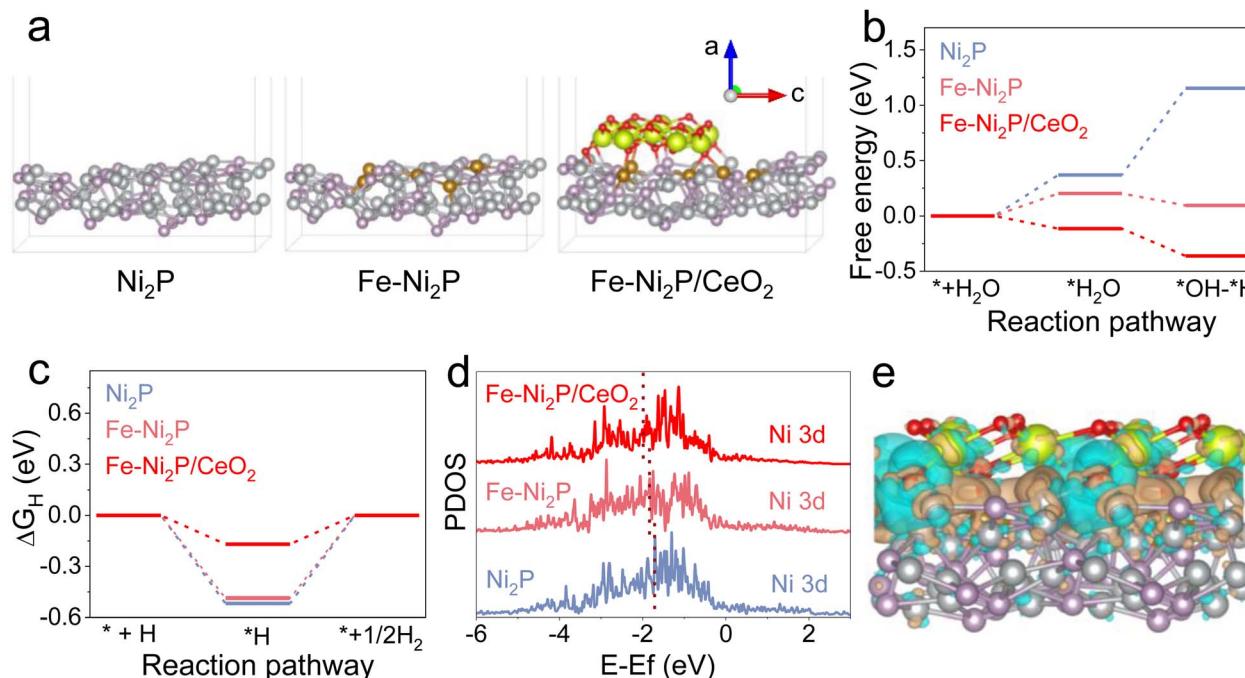


Fig. 6 (a) Structural models of Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂. Energy differences for (b) water dissociation and (c) ΔG_{H^*} of Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂. (d) Density of states of as-constructed models. (e) Charge density difference at the interface of Fe-Ni₂P and CeO₂.

for Fe-Ni₂P/CeO₂ signify its fast reaction kinetics and high intrinsic catalytic performance, which guarantees an effective HER process and is consistent with experimental results. In Fig. 6d, the projected partial density of states (PDOS) of Ni 3d in Ni₂P, Fe-Ni₂P and Fe-Ni₂P/CeO₂ are analyzed, and the corresponding d-band centers are calculated to be -1.05 , -1.71 , -1.84 , and -1.99 eV, respectively. Based on the d-band theory, the downshifted d-band center indicates increased filling of the as-hybridized antibonding orbitals, leading to fluctuation in the interaction between electronic structure and adsorbates on the surface of catalysts.⁴⁸ Therefore, Fe-Ni₂P/CeO₂ displays a small activity barrier to form chemical bonds and a low adsorption energy, which is conducive to the desorption of adsorption intermediates. The charge density difference of Fe-Ni₂P/CeO₂ (Fig. 6e) displays electron localization behavior and charge depletion, indicating the presence of charge transfer process on the interface of Fe-Ni₂P and CeO₂. These results confirm that the introduction of Fe and modification with CeO₂ can synergistically regulate the electronic structure of Ni₂P, thus accelerating the intrinsic catalytic kinetics of Fe-Ni₂P/CeO₂.

Conclusions

In summary, Fe-Ni₂P/CeO₂ nanosheet arrays were successfully constructed as a bifunctional catalyst to obtain energy-saving hydrogen generation by coupling HER with HzOR in alkaline seawater. The as-synthesized Fe-Ni₂P/CeO₂ reveals excellent catalytic performances toward the HER and HzOR due to the synergistic effect of Fe doping and CeO₂ coupling. Impressively, the integrated Fe-Ni₂P/CeO₂ couple in a two-electrode OHzS electrolyzer delivers small cell voltages of 0.051 and 0.597 V to

attain 10 and 400 mA cm⁻² and good stability at a large industrial current density of 400 mA cm⁻² with avoidable chlorine electrochemistry. The DFT calculations offer a theoretical analysis of the splendid catalytic performances of Fe-Ni₂P/CeO₂, in which the Fe doping and heterointerfaces between Fe-Ni₂P and CeO₂ can effectively modify the electronic structure and reduce the reaction barrier.

Data availability

The relevant experimental and characterization data are available in the article and the ESI.†

Author contributions

Rui-Qing Li: conceptualization, formal analysis, writing – review & editing. Songyun Guo: data curation. Xiaojun Wang: formal analysis. Xiaoyu Wan: data curation, formal analysis. Shuxiang Xie: investigation. Yu Liu: formal analysis. Changming Wang: investigation. Guangyu Zhang: methodology. Jun Cao: software. Jiamu Dai: formal analysis. Mingzheng Ge: methodology. Wei Zhang: conceptualization, writing – review & editing.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work is financially supported by the National Natural Science Foundation of China (No. 22302103), the Natural



Science Foundation of Jiangsu Province (No. BK20230619), the Natural Science Foundation of the Jiangsu Higher Education Institutions of China (No. 23KJB540003) and the Science Research Program of Nantong University (No. 03083110).

Notes and references

- J. Turner, *Nature*, 2004, **305**, 972.
- M. Chatenet, B. Pollet, D. Dekel, F. Dionigi, J. Deseure, P. Millet, R. Braatz, M. Bazant, M. Eikerling, I. Staffell, P. Balcombe, Y. Shao-Horn and H. Schäfer, *Chem. Soc. Rev.*, 2022, **51**, 4583.
- Y. Zuo, S. Bellani, G. Saleh, M. Ferri, D. Shinde, M. Zappia, J. Buha, R. Brescia, M. Prato, R. Pascazio, A. Annamalai, D. Souza, L. Trizio, I. Infante, F. Bonaccorso and L. Manna, *J. Am. Chem. Soc.*, 2023, **145**, 21419.
- X. Deng, X. Zheng, Z. Gong, W. Tan and X. Pei, *Chin. J. Rare Met.*, 2023, **47**, 43.
- Z. He, C. Zhang, S. Guo, P. Xu, Y. Ji, S. Luo, X. Qi, Y. Liu, N. Cheng, S. Dou, Y. Wang and B. Zhang, *Chem. Sci.*, 2024, **15**, 1123.
- H. Liu, W. Shen, H. Jin, J. Xu, P. Xi, J. Dong, Y. Zheng and S. Qiao, *Angew. Chem., Int. Ed.*, 2023, **62**, e2023116.
- J. Chang, G. Wang, Z. Yang, B. Li, Q. Wang, R. Kulieev, N. Orlovskaya, M. Gu, Y. Du, G. Wang and Y. Yang, *Adv. Mater.*, 2021, **33**, 2101425.
- Q. Liu, Z. Yan, J. Gao, H. Fan, M. Li and E. Wang, *Chem. Sci.*, 2023, **14**, 11830.
- X. Liu, Y. Jiang, J. Huang, W. Zhong, B. He, P. Jin and Y. Chen, *Carbon Energy*, 2023, **5**, e367.
- J. Chi, L. Guo, J. Mao, T. Cui, J. Zhu, Y. Xia, J. Lai and L. Wang, *Adv. Funct. Mater.*, 2023, **33**, 2300625.
- B. Sang, Y. Liu, X. Wan, S. Xie, G. Zhang, M. Ge, J. Dai, W. Zhang and R. Li, *Chem. Commun.*, 2023, **59**, 8743.
- W. Wang, X. Yu, H. He, Y. Wang, Y. Li, L. Deng and Y. Liu, *Chem. Eng. J.*, 2023, **465**, 142865.
- Z. Wang, Q. Hong, X. Wang, H. Huang, Y. Chen and S. Li, *Acta Phys.-Chim. Sin.*, 2023, **39**, 2303028.
- R. Li, S. Zeng, B. Sang, C. Xue, K. Qu, Y. Zhang, W. Zhang, G. Zhang, X. Liu, J. Deng, O. Fontaine and Y. Zhu, *Nano Res.*, 2023, **16**, 2543.
- Y. Zhu, J. Zhang, Q. Qian, Y. Li, Z. Li, Y. Liu, C. Xiao, G. Zhang and Y. Xie, *Angew. Chem., Int. Ed.*, 2022, **61**, e202113082.
- J. Kong, J. Zheng, H. Wu, C. Wan, M. Ye and L. Xu, *Chin. J. Rare Met.*, 2023, **47**, 797.
- Y. Yang, X. Li, G. Liu, H. Liu, Y. Shi, C. Ye, Z. Fang, M. Ye and J. Shen, *Adv. Mater.*, 2023, **36**, 2307979.
- H. Wang, M. Sun, J. Ren and Z. Yuan, *Adv. Energy Mater.*, 2023, **13**, 2203568.
- X. Liu, J. Chi, H. Mao and L. Wang, *Adv. Energy Mater.*, 2023, **13**, 2301438.
- Y. Zhang, R. Yan, X. Xu, C. He, M. Zhou, Z. Zeng, T. Ma, M. Wang and S. Li, *Adv. Funct. Mater.*, 2023, **34**, 2308813.
- Z. Ge, Y. Ding, T. Wang, F. Shi, P. Jin, P. Chen, B. He, S. Yin and Y. Chen, *J. Energy Chem.*, 2023, **77**, 209.
- R. Li, X. Wan, B. Chen, R. Cao, Q. Ji, J. Deng, K. Qu, X. Wang and Y. Zhu, *Chem. Eng. J.*, 2021, **409**, 128240.
- R. Li, C. Wang, S. Xie, T. Hang, X. Wan, J. Zeng and W. Zhang, *Chem. Commun.*, 2023, **59**, 11512.
- X. Zhang, D. Wu, X. Liu, Y. Qiu, Z. Liu, H. Xie, J. Duan and B. Hou, *Appl. Catal., B*, 2023, **330**, 122594.
- R. Li, B. Wang, T. Gao, R. Zhang, C. Xu, X. Jiang, J. Zeng, Y. Bando, P. Hu, Y. Li and X. Wang, *Nano Energy*, 2019, **58**, 870.
- H. Yang, P. Guo, R. Wang, Z. Chen, H. Xu, H. Pan, D. Sun, F. Fang and R. Wu, *Adv. Mater.*, 2022, **34**, 2107548.
- K. Zhang, G. Zhang, Q. Ji, J. Qu and H. Liu, *Nano-Micro Lett.*, 2020, **12**, 154.
- P. Bodhankar, P. Sarawade, P. Kumar, A. Vinu, A. Kulkarni, C. Lokhande and D. Dhawale, *Small*, 2023, **18**, 2107572.
- B. Sang, Y. Liu, X. Wan, S. Xie, G. Zhang, M. Ge, J. Dai, W. Zhang and R. Li, *Chem. Commun.*, 2023, **59**, 8743.
- S. Chu, W. Chen, G. Chen, J. Huang, R. Zhang, C. Song, X. Wang, C. Li and K. Ostrikov, *Appl. Catal., B*, 2019, **243**, 537.
- M. Ning, Y. Wang, L. Wu, L. Yang, Z. Chen, S. Song, Y. Yao, J. Bao, S. Chen and Z. Ren, *Nano-Micro Lett.*, 2023, **15**, 157.
- Q. Wang, R. He, F. Yang, X. Tian, H. Sui and L. Feng, *Chem. Eng. J.*, 2023, **456**, 141056.
- Y. Liu, J. Zhang, Y. Li, Q. Qian, Z. Li, Y. Zhu and G. Zhang, *Nat. Commun.*, 2020, **11**, 1853.
- C. Li, M. Chen, Y. Xie, H. Wang and L. Jia, *J. Colloid Interface Sci.*, 2023, **636**, 103.
- S. Meng, M. Sun, P. Zhang, C. Zhou, C. He, H. Zhang, Y. Liu, Z. Xiong, P. Zhou and B. Lai, *Environ. Sci. Technol.*, 2023, **57**, 12534.
- C. Yang, Y. Lu, L. Zhang, Z. Kong, T. Yang, L. Tao, Y. Zou and S. Wang, *Small Struct.*, 2021, **2**, 2100058.
- L. Wen, X. Zhang, J. Liu, X. Li, C. Xing, X. Lyu, W. Cai, W. Wang and Y. Li, *Small*, 2019, **15**, 1902373.
- S. Lee, L. Bai and X. Hu, *Angew. Chem., Int. Ed.*, 2020, **132**, 8149.
- H. Sun, C. Tian, G. Fan, J. Qi, Z. Liu, Z. Yan, F. Cheng, J. Chen, C. Li and M. Du, *Adv. Funct. Mater.*, 2020, **30**, 1910596.
- P. Zhou, G. Hai, G. Zhao, R. Li, X. Huang, Y. Lu and G. Wang, *Appl. Catal., B*, 2023, **325**, 122364.
- J. Ren, Y. Wang, L. Chen, L. Gao and W. Tian, *Chem. Eng. J.*, 2020, **389**, 124408.
- Q. He, Y. Zhou, H. Shou, X. Wang, P. Zhang, W. Xu, S. Qiao, C. Wu, H. Liu, D. Liu, S. Chen, R. Long, Z. Qi, X. Wu and L. Song, *Adv. Mater.*, 2022, **34**, 2110604.
- X. Zhao, K. Tang, C. Lee, C. Du, H. Yu, X. Wang, W. Qi, Q. Ye and Q. Yan, *Small*, 2022, **18**, 2107541.
- M. Luo, J. Yang, X. Li, M. Eguchi, Y. Yamauchi and Z. Wang, *Chem. Sci.*, 2023, **14**, 3400.
- S. Lu, C. Cheng, Y. Shi, Y. Wu, Z. Zhang and B. Zhang, *Proc. Natl. Acad. Sci. U. S. A.*, 2023, **120**, e2300549120.
- J. Li, J. Stenlid, T. Ludwig, P. Lamoureux and F. Abild-Pedersen, *J. Am. Chem. Soc.*, 2021, **143**, 19341.
- S. Zhang, C. Tan, R. Yan, X. Zou, F. Hu, Y. Mi, C. Yan and S. Zhao, *Angew. Chem., Int. Ed.*, 2023, **62**, e2023027.
- Z. Wang, C. Zhu, H. Tan, J. Liu, L. Xu, Y. Zhang, Y. Liu, X. Zou, Z. Liu and X. Lu, *Adv. Funct. Mater.*, 2021, **31**, 2104735.

