



reduced back into  $\text{Ni}^{2+}$ , thus realizing a reversible cycle of  $\text{Ni}^{2+}/\text{Ni}^{3+}$  in the MOR.<sup>14</sup> Unfortunately, the further development of Ni-based electrocatalysts in methanol economy is restricted due to the following three reasons. (1) The excess adsorption of intermediates during the MOR poisons Ni active sites,<sup>15</sup> thus limiting the continuous conversion from methanol to formate. (2) The overoxidation of Ni species, such as the transformation from  $\text{Ni}^{3+}$  to  $\text{Ni}^{4+}$  at high voltages, destroys the reversible  $\text{Ni}^{2+}/\text{Ni}^{3+}$  cycle, resulting in a dramatic decrease in activity.<sup>16</sup> (3) The current density of the MOR is still low, which is not suitable for industrial production ( $>300 \text{ mA cm}^{-2}$ ). Thus, developing durable Ni-based catalysts with high formate selectivity at high current densities is a big challenge.

It has been reported that  $\text{M}_a\text{--O--M}_b$  units with different metals can establish interface-bonding channels to enhance charge transfer and modulate the charge state of the surface adsorbate to activate reaction intermediates.<sup>17</sup> Considering the special characteristics of Cu, such as half-fully occupied 4s orbitals, a flexible hybrid spin state and abundant chemical valences,<sup>18,19</sup> we proposed the construction of Cu–O–Ni units. Among the latter, on the one hand, Cu serves as an electron donor to generate directional electron transfer from Cu to Ni through O atoms to lower the excess accumulation of intermediate  $\text{CO}_{\text{ad}}$  at Ni active sites. On the other hand, Cu inhibits the overoxidation of  $\text{Ni}^{3+}$  species during the MOR to stabilize the metastable high-valence  $\text{Ni}^{3+}$  active sites.<sup>20</sup> Besides, through the constructed Cu–O–Ni electron-transfer channel, the dehydrogenation ability of Ni is enhanced, leading to excellent MOR activity at high current density.

Specifically, we constructed a novel Cu–O–Ni unit to inhibit the overoxidation of Ni ( $\text{Ni}^{3+} \rightarrow \text{Ni}^{4+}$ ) by introducing Cu, in which the bimetallic oxide ( $\text{NiCuO}_x$ ) was loaded on nickel foam (NF) as an efficient MOR electrocatalyst *via* a two-step method involving electrochemical deposition and thermal treatment. The optimal  $\text{NiCuO}_x\text{-2/NF}$  electrocatalyst with an electrochemical deposition time of 2000 s exhibited excellent catalytic activity towards the MOR at an industrial-scale current density of  $500 \text{ mA cm}^{-2}$  with only 1.42 V (vs. RHE) and high faradaic efficiencies of  $\sim 100\%$ . Impressively, the  $\text{NiCuO}_x\text{-2/NF}$  catalyst could catalyze methanol oxidation stably up to 600 h at a high current density of  $\sim 500 \text{ mA cm}^{-2}$ , which is superior to reported Ni-based MOR catalysts. Based on XPS, electrochemical results and operando impedance measurements, we revealed that the introduction of Cu could generate an electron-transfer channel from Cu to Ni, which consequently promoted the  $\text{CH}_3\text{OH}^*$  dehydrogenation ability of Ni and stabilized  $\text{Ni}^{3+}$  active sites. Besides, the strong Cu–O–Ni interaction at oxide interfaces also avoids the overoxidation of Ni species ( $\text{Ni}^{3+} \rightarrow \text{Ni}^{4+}$ ) to a certain extent, making it possible to catalyze methanol effectively and stably at industrial-grade current densities ( $>300 \text{ mA cm}^{-2}$ ). By coupling the MOR with a cathodic HER, the integrated two-electrode system using  $\text{NiCuO}_x\text{-2/NF}$  as both anode and cathode catalysts could reach a current density of  $400 \text{ mA cm}^{-2}$  at 2.02 V, which is lower than that of the electrolytic water splitting by  $\sim 230 \text{ mV}$ , thereby consolidating the application potential in commercial  $\text{H}_2$  production and biomass increment conversion.

## Results and discussion

### Synthesis and characterizations

$\text{NiCuO}_x$  nanostructures were grown on NF by a two-step method based on electrochemical deposition, followed by further heat treatment (Fig. 1a). First, the pretreated NF was immersed and successively deposited electrochemically in  $\text{H}_2\text{SO}_4$  solutions with  $\text{NiCl}_2\cdot 6\text{H}_2\text{O}$  and  $\text{CuCl}_2\cdot 6\text{H}_2\text{O}$  as Ni and Cu sources, respectively. Then, the resultant  $\text{NiCu/NF}$  precursor was heated slightly in a muffle furnace to obtain  $\text{NiCuO}_x\text{/NF}$ . To study the effect of different amounts of Cu doping,  $\text{NiCuO}_x$  samples were prepared by changing the electrodeposition time from 1000 s to 4000 s in  $\text{CuCl}_2$  solutions.

Scanning electron microscopy (SEM) images (Fig. 1b) showed that  $\text{NiCuO}_x\text{-2/NF}$  with an electrodeposition time of 2000 s consisted of both nanoneedles and cubic nanoparticles, which covered the surface of NF densely and uniformly.  $\text{NiCuO}_x\text{-1/NF}$  (Fig. S1†) showed a similar structure to  $\text{NiCuO}_x\text{-2/NF}$ , but was not fully grown due to the insufficient electrodeposition time. As the electrodeposition time increased further, the nanoneedles gradually disappeared and only cubic and agglomerated nanoparticles were observed for  $\text{NiCuO}_x\text{-3/NF}$  (Fig. S2†) compared with the flat or irregular morphology of control samples of  $\text{CuO}_x\text{/NF}$  (Fig. S3†) and  $\text{Ni/NF}$  (Fig. S4†). Furthermore, the needle-like and cubic morphologies were also confirmed by transmission electron microscopy (TEM) images (Fig. 1c). The TEM images (Fig. S5 and S6†) of control samples showed similar evolution trends to those in the SEM images. The lattice spacings of  $2.41 \text{ \AA}$  and  $2.08 \text{ \AA}$  in high-resolution TEM (HRTEM) images (Fig. 1d and e) corresponded with the (111) plane of  $\text{NiCuO}_x$  and (111) plane of NiCu alloy, respectively. It has been reported that the (111) crystal planes of Cu and Ni can enhance the MOR activity of catalysts. Thus, the exposure of (111) crystal faces in  $\text{NiCuO}_x\text{/NF}$  facilitated the adsorption and dissociation of methanol molecules and enhanced the high MOR electrocatalytic activity. Besides, the high-angle annular dark field (HAADF) image and corresponding energy dispersive spectroscopy (EDS) mappings demonstrated the uniform distribution of Ni, Cu and O elements in both nanoneedles and nanoparticles of  $\text{NiCuO}_x\text{-2/NF}$  (Fig. 1f–i). The content ratio of Ni to Cu in  $\text{NiCuO}_x$  gradually decreased with the longer electrodeposition time of Cu source, and the specific data are displayed in Tables S1 and S2.†

X-ray diffraction (XRD) patterns were recorded to identify the crystalline structure of as-prepared catalysts.  $\text{Ni/NF}$  showed three obvious characteristic peaks with  $2\theta$  of  $44.5^\circ$ ,  $51.8^\circ$  and  $76.4^\circ$  corresponding to cubic-phase Ni (PDF#87-0712) (Fig. S7†).<sup>21</sup> After introduction of the Cu source (Fig. 2a), the diffraction peaks at  $43.5^\circ$  and  $44.6^\circ$  could be assigned to the cubic phases of NiCu alloy and  $\text{NiCuO}_x$  oxide, respectively. The slight shifts to a low angle of the corresponding characteristic peaks (PDF#85-1326, PDF#73-1519) for  $\text{NiCuO}_x$ , as well as the elemental mapping analysis stated above confirmed the generation of Cu–Ni lattice mixing. The atom substitution of Cu to Ni for pristine  $\text{NiO}$  lattice led to the generation of Cu–O–Ni units. Besides, the intensity ratio of  $\text{NiCuO}_x$  to NiCu increased



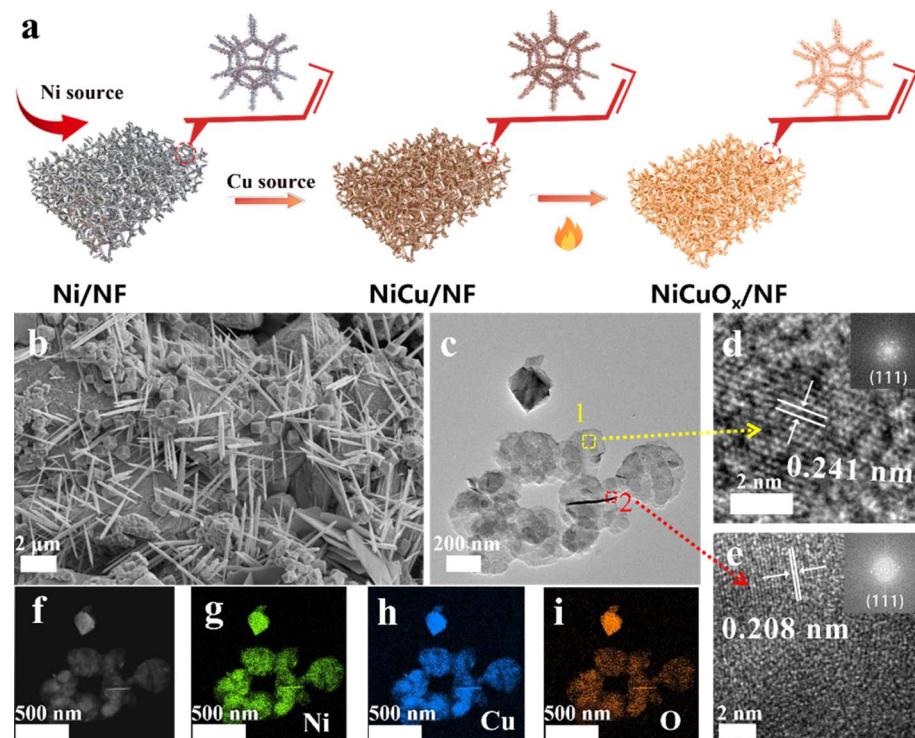


Fig. 1 (a) Synthesis of  $\text{NiCuO}_x/\text{NF}$  (schematic). (b) SEM, (c) TEM, (d and e) HRTEM images and (f–i) EDS elemental mappings of  $\text{NiCuO}_x\text{-2/NF}$ .

with a longer electrodeposition time, revealing that the introduction of Cu could facilitate the generation of oxidized Ni species. The diffraction peaks at  $28.7^\circ$  and  $47.6^\circ$  attributed to cubic  $\text{CuCl}$  (PDF#77-2383) disappeared after the electrochemical test. The XRD pattern of  $\text{CuO}_x/\text{NF}$  (Fig. S8†) showed similar diffraction peaks to those of  $\text{NiCuO}_x$ .

The chemical compositions and oxidation states of as-synthesized catalysts were probed using the X-ray

photoelectron spectroscopy (XPS). As shown in Fig. 2b, Cu, Ni, O, C and Cl elements could be observed for  $\text{NiCuO}_x\text{-1/NF}$ ,  $\text{NiCuO}_x\text{-2/NF}$  and  $\text{NiCuO}_x\text{-3/NF}$ , data which were consistent with XRD results.

Fig. 2c shows the XPS Ni 2p spectra for  $\text{Ni}/\text{NF}$ ,  $\text{NiCuO}_x\text{-1/NF}$ ,  $\text{NiCuO}_x\text{-2/NF}$  and  $\text{NiCuO}_x\text{-3/NF}$ .  $\text{Ni}/\text{NF}$  mainly demonstrated the metal  $\text{Ni}^0$  with corresponding binding energies at about 853 and 870.2 eV assigned to  $\text{Ni}^0 2\text{p}_{3/2}$ , and  $\text{Ni}^0 2\text{p}_{1/2}$ , respectively.

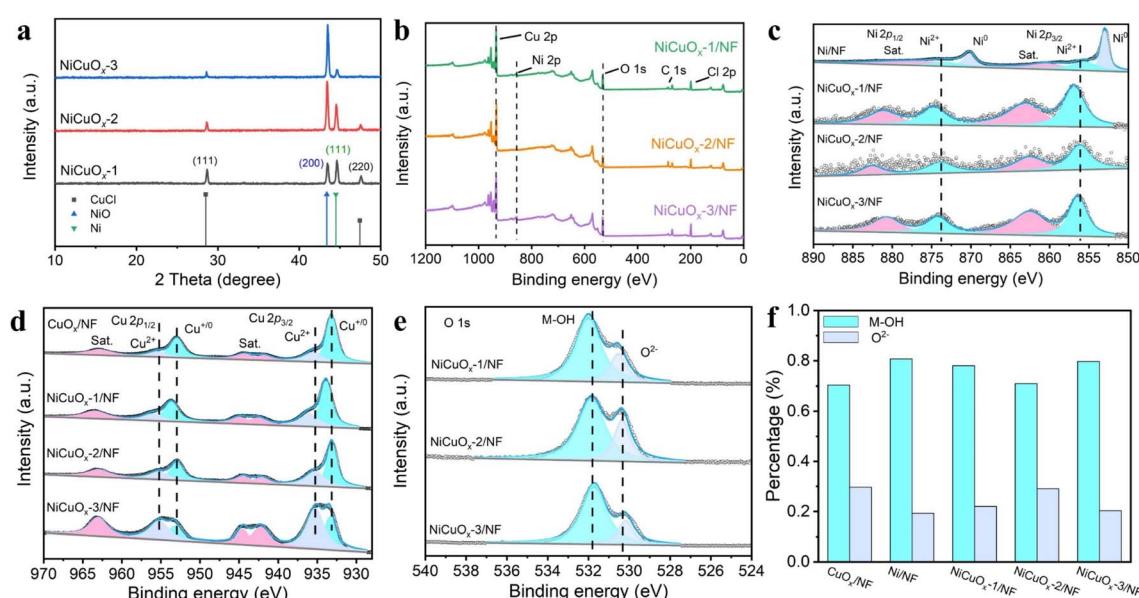


Fig. 2 (a) XRD patterns for  $\text{NiCuO}_x\text{-1}$ ,  $\text{NiCuO}_x\text{-2}$  and  $\text{NiCuO}_x\text{-3}$ . Survey XPS spectrum (b), high-resolution XPS Ni 2p (c), Cu 2p (d), O 1s (e) and corresponding histograms (f) of  $\text{M-OH}/\text{O}^{2-}$  sub-bands for as-prepared catalysts.

After Cu introduction, the chemical valence of Ni in  $\text{NiCuO}_x/\text{NF}$  composites increased to Ni(II), and the binding energies at 856.1 and 873.3 eV were attributed to  $\text{Ni}^{2+} 2p_{3/2}$  and  $\text{Ni}^{2+} 2p_{1/2}$ , respectively.<sup>22</sup> Among  $\text{NiCuO}_x/\text{NF}$  composites,  $\text{NiCuO}_x\text{-2/NF}$  exhibited a slightly negative shift in Ni 2p spectra, indicating the local electronic transfer from Cu to Ni. The Cu 2p spectra in Fig. 2d display the characteristic sub-bands at the binding energies of 933.2, 935.2, 953 and 955.2 eV, which were assigned to  $\text{Cu}^{+0} 2p_{3/2}$ ,  $\text{Cu}^{2+} 2p_{3/2}$ ,  $\text{Cu}^{+0} 2p_{1/2}$  and  $\text{Cu}^{2+} 2p_{1/2}$ , respectively.<sup>23</sup> The specific ratios of  $\text{Cu}^{2+}$  to  $\text{Cu}^{+0}$  in the material gradually increased as the amount of Cu loading increased, which is consistent with the result of Ni 2p XPS spectra. Further, the O 1s spectra (Fig. 2e, S10 and S11†) of as-prepared catalysts were deconvoluted into two sub-bands at  $\sim 530.4$  and  $\sim 531.8$  eV, which corresponded to lattice oxygen (metal-bonded  $\text{O}^{2-}$ ) and surface-chemisorbed oxygen (oxygen in M-OH), respectively.<sup>24</sup>  $\text{NiCuO}_x\text{-2}$  showed the highest proportion of  $\text{O}^{2-}$  (Fig. 2f), which is beneficial for lattice stability under high current densities.

During the MOR,  $\text{Ni}^{2+}$  species in Ni-based materials tend to be *in situ* electrochemically oxidized to form active  $\text{Ni}^{3+}$  with high MOR catalytic activity. Unfortunately, as the current density increases to industrial grade, active  $\text{Ni}^{3+}$  can undergo further oxidation, leading to a reduction in catalytic activity. According to the XPS results, the introduction of Cu in  $\text{NiCuO}_x\text{-2/NF}$  to construct a Cu-O-Ni unit could stabilize the lattice and prevent the further oxidation of  $\text{Ni}^{3+}$  during the MOR, leading to the ultra-long MOR stability of  $\text{NiCuO}_x\text{-2/NF}$ .

### Electrocatalytic performance evaluation

The MOR performances of as-prepared catalysts were investigated in 1 M KOH solution with different concentrations of MeOH using linear sweep voltammetry (LSV). First, the onset oxidation potentials of the MOR (Fig. 3a) for  $\text{NiCuO}_x\text{-2/NF}$  decreased significantly after the addition of MeOH to the electrolyte. The MOR current density increased significantly with increasing of the concentration of MeOH up to 1 M. As the concentration increased to 2 M, the current density for the MOR began to decrease owing to excess adsorption of methanol molecules. Therefore, 1 M of MeOH was chosen as the additive concentration for subsequent MOR measurements. The cyclic voltammetry (CV) curves in Fig. 3b suggested that all of the catalysts had one pair of reversible redox peaks corresponding to the  $\text{Ni}^{2+}/\text{Ni}^{3+}$  transformation.<sup>25</sup> The sharp increase in current densities above  $\sim 1.2$  V matched well with the Ni(II)  $\rightarrow$  Ni(III) OOH oxidation, and the backward scanning peak of  $\text{NiCuO}_x/\text{NF}$  corresponding to Ni(III) reduction shifted to the lower potential compared with Ni/NF. More impressively, the lower reduction potential of Ni(III)  $\rightarrow$  Ni(II) in  $\text{NiCuO}_x\text{-2/NF}$  was responsible to the higher current density and better MOR performance.

The LSV curves in Fig. 3c suggested that  $\text{NiCuO}_x\text{-2/NF}$  exhibited the best MOR performance in as-prepared catalysts.  $\text{NiCuO}_x\text{-2/NF}$  needed only 1.42 V to reach an industrial-scale current density of 500 mA cm<sup>-2</sup>, which was lower than that of Ni/NF,  $\text{CuO}_x/\text{NF}$ ,  $\text{NiCuO}_x\text{-1/NF}$  and  $\text{NiCuO}_x\text{-3/NF}$  by 153, 104, 50 and 16 mV, respectively. This result indicated that the amount of Cu species introduced in  $\text{NiCuO}_x\text{-2/NF}$  with electrochemical

deposition of 2000 s was optimal for the MOR performance, and that excessive Cu species would cover the catalyst surface and inhibit the oxidation of Ni(II) to active Ni(III). The onset potentials ( $\sim 1.3$  V) of three  $\text{NiCuO}_x/\text{NF}$  species were close to that of the Ni(II)/Ni(III) redox couple, confirming that Ni(III) was the active species for the MOR. Moreover, the Tafel slope of  $\text{NiCuO}_x\text{-2/NF}$  was 87.2 mV dec<sup>-1</sup> (Fig. 3d), which was much lower than that of  $\text{CuO}_x/\text{NF}$  (100 mV dec<sup>-1</sup>), Ni/NF (116.1 mV dec<sup>-1</sup>),  $\text{NiCuO}_x\text{-1/NF}$  (91.1 mV dec<sup>-1</sup>) and  $\text{NiCuO}_x\text{-3/NF}$  (107.2 mV dec<sup>-1</sup>), implying the fast MOR kinetics of  $\text{NiCuO}_x\text{-2/NF}$ . Fig. 3e reveals the electrochemical impedance spectroscopy (EIS) data. The resistance to electron transfer increased in the order Ni/NF  $>$   $\text{NiCuO}_x\text{-1/NF}$   $>$   $\text{NiCuO}_x\text{-3/NF}$   $>$   $\text{NiCuO}_x\text{-2/NF}$ , which was in good agreement with the LSV results stated above. The lower resistance and faster charge transport/diffusion ability of  $\text{NiCuO}_x\text{-2/NF}$  contributed to the more significant enhancement of MOR catalytic activity.

Furthermore, <sup>1</sup>H and <sup>13</sup>C nuclear magnetic resonance (NMR) spectroscopy were carried out to identify the products of MOR. Formate was the main product after chronoamperometric (*i-t*) tests for 1 h (Fig. S12†), and the faradaic efficiency (FE) for methanol-to-formate oxidation at different current densities was also investigated. The FE of MOR for  $\text{NiCuO}_x\text{-2/NF}$  was  $\sim 100\%$  at different current densities and  $> 96.8\%$  even at a high current density of 500 mA cm<sup>-2</sup>, suggesting the ultrahigh selectivity of  $\text{NiCuO}_x\text{-2/NF}$  (Fig. 3f). Furthermore, the formate concentrations generated by the  $\text{NiCuO}_x\text{-2/NF}$  anode at different current densities were investigated. The average generation rate of formate was 83.09, 107.14, 148.75 and 184.03 mmol cm<sup>-2</sup> h<sup>-1</sup> at a current density of 200, 300, 400 and 500 mA cm<sup>-2</sup>, respectively.

Specifically, the durability of the operation and prospect of industrial application of  $\text{NiCuO}_x\text{-2/NF}$  was estimated by chronoamperometry at an especially high current density ( $\sim 500$  mA cm<sup>-2</sup>) under 1.63 V (vs. RHE, without *iR*-compensation). Encouragingly, the current density did not experience severe degradation after the stability test of up to 600 h, indicating the superior stability of  $\text{NiCuO}_x\text{-2/NF}$  for the MOR. After the stability test, there was no obvious change in the main crystalline phases of NiCu alloy and  $\text{NiCuO}_x$  in  $\text{NiCuO}_x\text{-2/NF}$  except for the disappearance of the diffused NiCl phase (Fig. S13†). The slight shift to a low angle was attributed to the introduction of more Cu in the  $\text{NiCuO}_x$  lattice. The emergence of high-valence  $\text{Cu}^{2+}$  and  $\text{Cu}^+$  peaks in Raman spectrum (Fig. S14†) suggested a deeper alloying effect of Cu during the MOR.<sup>33</sup> SEM and TEM images of post-testing  $\text{NiCuO}_x\text{-2/NF}$  showed a morphology of nanoplates, and no impurity phase was observed (Fig. S15 and S16†). Besides, the XPS results after the stability test (Fig. S17†) showed that high-valence  $\text{Ni}^{4+}$  species were not formed in  $\text{NiCuO}_x\text{-2/NF}$ . Hence, the formation of  $\text{Ni}^{4+}$  was suppressed by the introduction of Cu species. The addition of Cu as the charge-compensation agent could inhibit the overoxidation of Ni and promote lattice stability for the  $\text{NiCuO}_x\text{-2/NF}$  catalyst. Compared with reported self-supporting nickel-based MOR catalysts, the designed  $\text{NiCuO}_x\text{-2/NF}$  obtained in our study was extremely competitive and showed superior stability for the



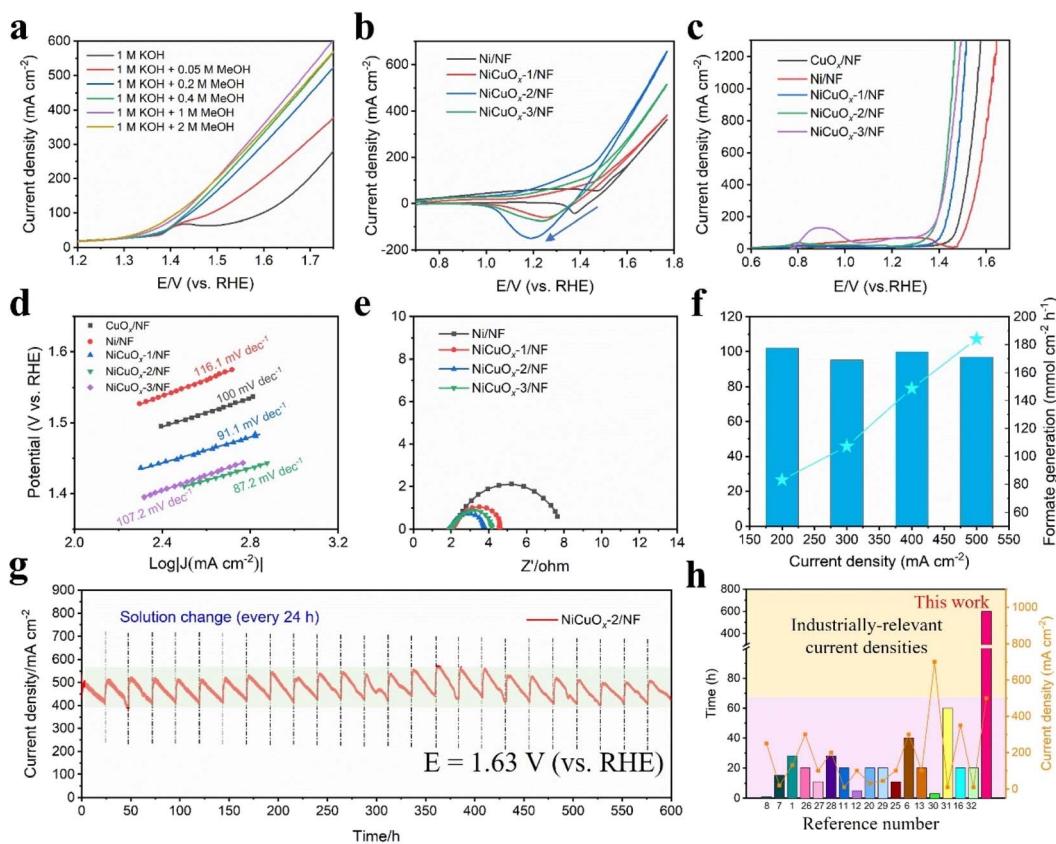


Fig. 3 (a) Polarization curves of NiCuO<sub>x</sub>-2/NF at different concentrations of methanol in KOH (1 M). (b) CV curves, (c) LSV curves and (d) corresponding Tafel plots of samples (with 100% *iR* compensation). (e) Nyquist plots measured in KOH (1 M) + MeOH (1 M). (f) FE and averaged generation rates of formate at the anode. (g) Chronoamperometry (*I*-*t*) curves of NiCuO<sub>x</sub>-2/NF at a constant potential of 1.63 V (vs. RHE). (h) Comparisons of test duration and corresponding current density between this work and catalysts reported in the literature.<sup>26–32</sup>

longest period of time at industrial-grade current densities (Fig. 3h and Table S3†).

### Insight into the MOR mechanism

Operando electrochemical impedance spectroscopy (EIS) was employed to reveal the electrochemical mechanism of NiCuO<sub>x</sub>-2/NF during MOR under different potentials. In the low-frequency region (0.01–10 Hz), the peak is usually regarded as nonhomogeneous charge distribution associated with the surface oxidizing species on the electrode surface (OER/MOR).<sup>34,35</sup> First, from the Bodes plots of NiCuO<sub>x</sub>-2/NF (Fig. 4a and b), an apparent peak could be found at a potential of 1.25 V during the MOR, lower than that during the OER (1.5 V), indicating faster MOR kinetics than for the OER. The Nyquist plots of NiCuO<sub>x</sub>-2/NF (Fig. S18†) also showed a smaller semicircle, denoting the lower impedance and more rapid electronic transfer towards the MOR. Impressively, the peak at 1.25 V shifted slowly to a higher frequency and lower phase angle with increasing the potential to 1.50 V (Fig. 4c), which was because the reaction kinetics of methanol oxidation are superior to the OER under low potentials.<sup>5</sup> As the polarization potential changed to >1.50 V, the frequency peak decreased with increasing voltage, leading to a decrease in Faraday resistance and an increase in the surface reaction rate.

Furthermore, the MOR and OER measurements of multi-potential steps were performed on NiCuO<sub>x</sub>-2/NF to investigate the MOR active sites and catalytic mechanism (Fig. 4d). First, the initial applied potential for the NiCuO<sub>x</sub>-2/NF anode was set to 1.62 V (vs. RHE) to produce abundant Ni<sup>III</sup>-OOH species. Then, the applied potential was switched to 0.57 V (vs. RHE) after entering an open-circuit state for 90 s. Under OER conditions, NiCuO<sub>x</sub>-2/NF showed significant reduction currents at 0.57 V (vs. RHE), suggesting non-spontaneous reduction of Ni<sup>3+</sup> in the absence of methanol. In contrast, when methanol was added in the open-circuit state, only weak reduction currents appeared at 0.57 V (vs. RHE), suggesting that the addition of methanol promoted the transition of Ni<sup>3+</sup> to Ni<sup>2+</sup> under MOR conditions. In addition, when methanol was added prior to the measurement, higher current densities were observed at 1.62 V (vs. RHE) compared with the data obtained under OER conditions, confirming the occurrence of the MOR. After the open circuit, only a weak current density was detected at 0.57 V (vs. RHE), indicating that the generated Ni<sup>(III)</sup> had been consumed simultaneously under MOR conditions. Therefore, we concluded that the MOR process of NiCuO<sub>x</sub>-2/NF involved the reversible Ni<sup>(II)</sup>/Ni<sup>(III)</sup> cycle (Fig. 4e): (i) the Ni<sup>(II)</sup> species in NiCuO<sub>x</sub>-2/NF are oxidized to high-valence Ni<sup>(III)</sup> under the oxidation potential; (ii) methanol is oxidized to formic acid by

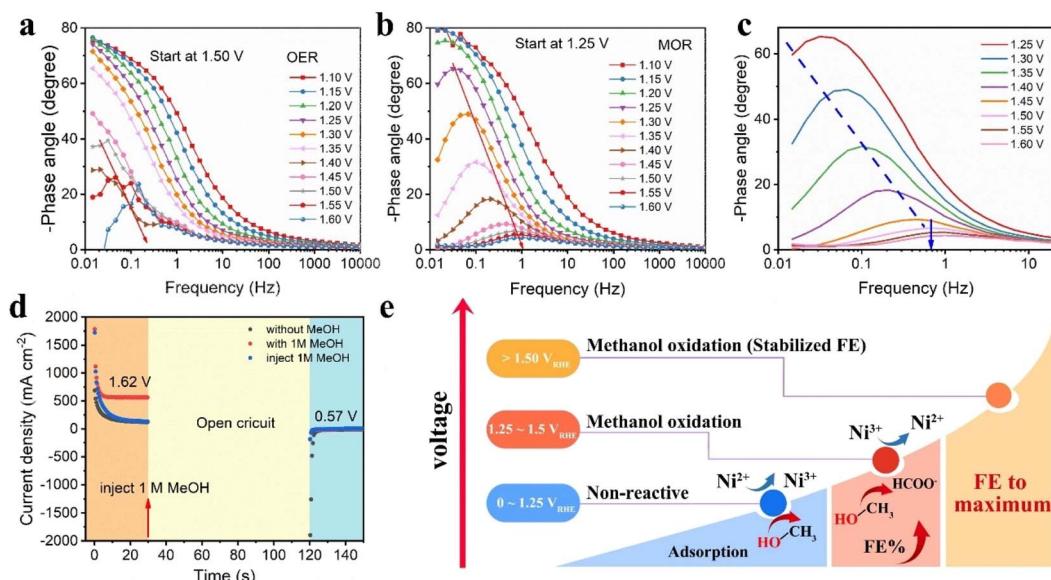


Fig. 4 Reaction mechanism of catalysts. (a)–(c) Bode plots of  $\text{NiCuO}_x\text{-2/NF}$  for the OER (a) and MOR (b) at different potentials. (d) Multi-potential step curves of  $\text{NiCuO}_x\text{-2/NF}$  in KOH solution (1 M) without and with MeOH (1 M). In all panels, the orange-, yellow-, and light blue-shaded areas indicate application of a constant voltage of 1.62 V (vs. RHE), an open-circuit process and a constant voltage of 0.57 V (vs. RHE), respectively. (e) The MOR reaction scheme at different potentials on  $\text{NiCuO}_x\text{-2/NF}$ .

the real active site Ni(III), and the accumulated Ni(III) is reduced back to initial Ni(II) concurrently. At the same time, the FE of methanol to formic acid increases steadily with increasing voltage and remains stable after reaching a maximum value until 1.50 V (vs. RHE), which agreed well with the result shown in Fig. 4c. Besides, through combining the electrochemical test results, the introduction of Cu could avoid the overoxidation of Ni species ( $\text{Ni}^{3+} \rightarrow \text{Ni}^{4+}$ ), thus maintaining the reversible Ni(II)/

Ni(III) cycle and realizing superior MOR activity, especially under high current densities.

#### Co-electrolysis in MeOH-water solution

The  $\text{NiCuO}_x\text{-2/NF}$  sample also showed HER catalytic ability in KOH (1 M) with/without containing methanol (1 M) (Fig. S19†). Hence, a MeOH–water solution electrolyser in the two-electrode system was constructed with  $\text{NiCuO}_x\text{-2/NF}$  simultaneously used

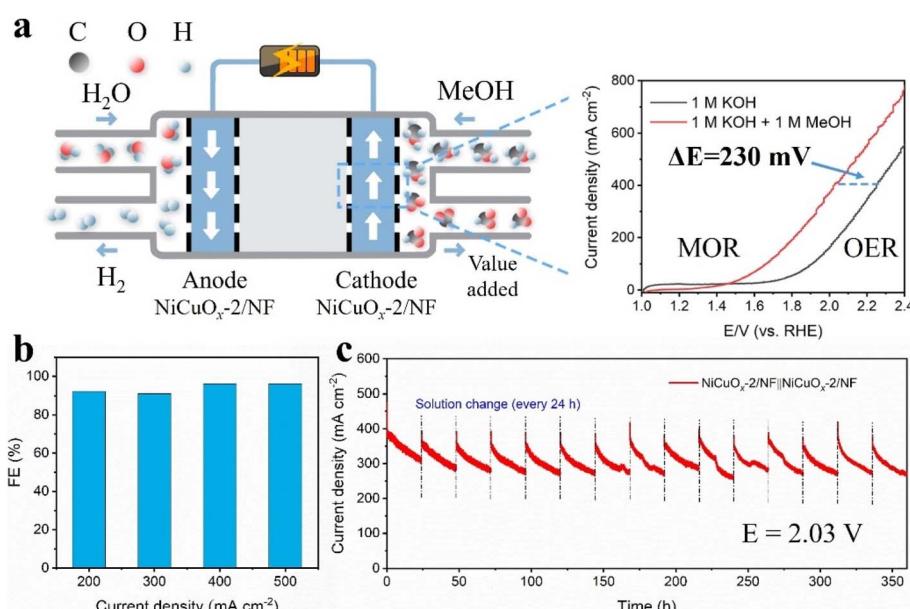


Fig. 5 (a) Scheme for a two-electrode electrolyzer for concurrent electrolytic production of hydrogen and value-added chemicals, and LSV curves of  $\text{NiCuO}_x\text{-2/NF}$  in KOH (1.0 M) with and without addition of methanol (1 M). (b) FE% of generated formate at the anode after operating at 200, 300, 400 and 500  $\text{mA cm}^{-2}$  for 1 h. (c) Chronoamperometry ( $I$ – $t$ ) curves by  $\text{NiCuO}_x\text{-2/NF}$ -equipped MOR||HER at a constant potential of 2.03 V.

as the catalyst at both the cathode and anode. As exemplified in Fig. 5a, the MOR occurred at the anode to produce high value-added products, and the HER occurred at the cathode to generate  $H_2$ . To reach a current density of  $400 \text{ mA cm}^{-2}$ , the MeOH–water electrolysis cell required a cell voltage of only 2.02 V. This was lower (by  $\sim 230 \text{ mV}$ ) than that in the electrolyte without MeOH (2.25 V), suggesting a significantly reduced electric consumption by coupling the MOR with the HER and valued chemical productions at the anode simultaneously.

In order to determine the oxidation products and calculate the corresponding FE, the MeOH–water electrolysis cell catalyzed by  $\text{NiCuO}_x\text{-2/NF}$  catalysts was operated at a current density of 200, 300, 400, and  $500 \text{ mA cm}^{-2}$ , respectively, and the liquid samples were collected after running for 1 h. It is noteworthy that FE could be maintained  $>96\%$  at high current densities, even at 400 and  $500 \text{ mA cm}^{-2}$  (Fig. 5b). According to the long-term chronoamperometry test at 2.03 V (Fig. 5c), the  $\text{NiCuO}_x\text{-2/NF}$  equipped two-electrode system could stand for 360 h without an obvious decrease in current density. Hence, it could find application as a bifunctional electrocatalyst in  $H_2$  production and biomass increment conversion.

## Conclusions

A novel Cu–O–Ni charge-transfer channel was developed in  $\text{NiCuO}_x$  bimetallic oxide to boost MOR electrocatalytic activity. The as-prepared  $\text{NiCuO}_x\text{-2/NF}$  catalyst exhibited an excellent electrocatalytic MOR performance. As low as 1.42 V (vs. RHE) was needed to achieve a high current density of  $500 \text{ mA cm}^{-2}$ . FE of  $\sim 100\%$  for formate generation and superior stability up to 600 h at industrial current densities were observed. The high MOR performance of  $\text{NiCuO}_x\text{-2/NF}$  was because the over-oxidation of active site Ni(III) could be effectively inhibited by the introduction of Cu through the formed Cu–O–Ni electron-transfer channel to maintain the reversible Ni(II)/Ni(III) transformation. On the other hand, the electronic transfer from Cu to Ni alleviated the accumulation of intermediate  $\text{CO}_{\text{ad}}$  at Ni sites, which led to the remarkable catalytic activity and stability at an increased current density. Besides, an established electrolytic cell created by coupling the MOR with the HER with  $\text{NiCuO}_x\text{-2/NF}$  as both anode and cathode catalysts outputs elicited a current density of  $400 \text{ mA cm}^{-2}$  at 2.02 V, which is  $\sim 230 \text{ mV}$  lower than that obtained by the splitting of pure water. This work provides a new perspective for the design of efficient and stable MOR electrocatalysts at high current density, and also promotes the development for energy-saving  $H_2$  production by coupling the oxidation of small molecules in biomass.

## Data availability

The data underlying this study are available in the published article and its ESI.†

## Author contributions

H. T.: data curation, investigation, methodology, data analysis, funding acquisition, and writing (original draft). X. W.: data

curation, investigation, and data analysis. W. L.: data curation and investigation. R. M. and X. Y.: data curation. S. L. and F. K.: investigation. X. C.: conceptualization, funding acquisition, supervision, and writing (review and editing). S. J.: conceptualization and supervision.

## Conflicts of interest

There are no conflicts of interest to declare.

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