

## EDGE ARTICLE

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## Introduction

Birefringence is manifested by the existence of orientation-dependent differences in the refractive index, which is widely used in the fields of optical polarization, optical modulation, and nonlinear optics.<sup>1–7</sup> The exploration of birefringent materials is an essential topic in the field of optical research.<sup>8–15</sup> To date, a diverse range of oxide and halide birefringent materials with high chemical stability have garnered significant interest, including  $MgF_2$ ,<sup>16</sup>  $\alpha$ - $BaB_2O_4$ ,<sup>17</sup>  $YVO_4$ ,<sup>18</sup>  $TiO_2$ ,<sup>19</sup>  $LiNbO_3$ ,<sup>20</sup>  $CaCO_3$ ,<sup>21</sup> etc. However, the aforementioned birefringent materials are plagued by various inherent defects:  $MgF_2$  possesses a tiny birefringence;  $\alpha$ - $BaB_2O_4$  encounters issues such as phase transition and cracking during growth;  $TiO_2$  demonstrates high hardness and poses challenges in terms of processing;  $YVO_4$  cannot be utilized within the ultraviolet band;  $LiNbO_3$  exhibits a low laser-induced damage threshold and is susceptible to photorefractive damage;  $CaCO_3$  is deliquescent and difficult to obtain high-quality single crystals. Hence, it holds immense significance for practical applications to discover birefringent materials with substantial birefringence, multi-wave

applicability, excellent chemical and physical stability, as well as favorable crystal growth habits.

Metal halides are highly regarded as important optical function materials due to their advantages of easy preparation, rich coordination environment, wide transparent range, high laser-induced damage threshold, and are applied in the frontier fields of luminescence, solar cells, laser frequency conversion, and so on.<sup>22–29</sup> Among them, binary metal halides are widely utilized owing to their simple composition and cost-effectiveness:  $KBr$  is commonly employed as a background material in Fourier transform infrared (FT-IR) spectroscopy due to its wide transparent range exceeding 25  $\mu m$ ,<sup>30</sup>  $CaF_2$  and  $BaF_2$  exhibit excellent mechanical properties, thermal stability, and radiation resistance along with high transparency from deep ultraviolet (UV) to IR regions, which are applied in optical prisms, lenses, wedge plates, diaphragms and other important optical components.<sup>31</sup> By reason of the foregoing, the splendid physical and chemical properties of binary metal halides are in line with our expectations for the next generation of birefringent crystal materials, making them regarded as a treasury of birefringent materials with great potential. On the other hand, metal halides display diverse coordination patterns, including linear, trigonal pyramidal, tetrahedral, and square pyramidal structures, which offer promising opportunities for identifying building blocks with significant polarizability anisotropy for birefringent materials.<sup>25</sup> Through comparison and screening, binary mercury-based (Hg-based) halides have emerged as a focal point for us due to their abundant configurations and broad transparent range. In Hg-based halides, apart from the traditional  $[HgX_4]$  ( $X$  = halogen) tetrahedra, there also exist infrequent  $[X-Hg-X]$  or  $[X-Hg-Hg-X]$  linear units.<sup>32–36</sup> In

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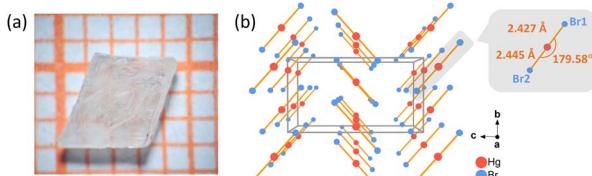


Fig. 1 (a) The single-crystal  $\text{HgBr}_2$  with a size of  $4 \times 3 \times 2 \text{ mm}^3$ ; (b) the structure of  $\text{HgBr}_2$  viewed along the  $a$ -axis (left) and the coordination environment of the  $\text{Hg}$  atom (right).

borates, the linear unit  $[\text{BO}_2]$  is thought to have greater polarizability anisotropy than the classic  $[\text{BO}_3]$ ,<sup>37,38</sup> and the previous studies of  $\text{HgB}_2\text{S}_4$  and trigonal  $\text{HgS}$  have suggested that the  $[\text{S}-\text{Hg}-\text{S}]$  linear units have strong birefringent contributions (0.28 at 1064 nm and 0.29 at 2100 nm, respectively).<sup>39,40</sup> Moreover, the  $[\text{X}-\text{Hg}-\text{Hg}-\text{X}]$  motif in the binary mercurous halide can induce a giant birefringence; however, prolonged exposure to light can lead to decomposition of mercurous halide.<sup>41</sup> Therefore, the linear  $[\text{X}-\text{Hg}-\text{X}]$  unit can also be regarded as a potential emerging building block of birefringent materials; however, its comprehensive investigation remains insufficient. Moreover, binary mercuric halides offer greater advantages in terms of synthesis and crystal growth, with their growth conditions generally being more moderate, thereby facilitating the achievement of large-scale crystal growth and device fabrication. Consequently, simple halides composed of linear  $[\text{X}-\text{Hg}-\text{X}]$  units are promising birefringent materials. In this work, mercuric bromide was selected as a potential birefringent material for the first time. A colorless single-crystal of mercuric bromide with a size of  $4 \times 3 \times 2 \text{ mm}^3$  was successfully obtained through a slow solvent evaporation technique employing hot ethanol as the solvent (as shown in Fig. 1a). The birefringence of  $\text{HgBr}_2$  at 546 nm is 0.235, which is 19.6 times higher than that of another commercial binary halide material,  $\text{MgF}_2$  (0.012 @ 546 nm).<sup>16</sup> Moreover,  $\text{HgBr}_2$  demonstrates a transparent range spanning from 0.34 to 25  $\mu\text{m}$ , effectively covering an ultrawide transparent range from UV to far-IR. The rationality of the experimental results is confirmed by theoretical calculations, indicating the potential of mercuric bromide as a next-generation birefringent material.

## Results and discussion

We have computationally determined the polarizability anisotropy and HOMO-LUMO gap of five Hg-based linear units (three  $[\text{HgX}_2]$ ,  $\text{X} = \text{Cl}, \text{Br}, \text{I}$ ; two  $[\text{HgQ}_2]$ ,  $\text{Q} = \text{S}, \text{Se}$ ), aiming to substantiate their potential as birefringent materials and analyze their respective merits and demerits. As depicted in Fig. 2a, the polarizability anisotropy of most Hg-based linear units is comparable to or even surpasses that of  $[\text{BS}_3]$ , the novel IR birefringent material building unit. Among them,  $[\text{HgX}_2]$  exhibits a wider HOMO-LUMO gap compared to  $[\text{BS}_3]$ , which is advantageous for achieving a wider band gap and a higher laser-induced damage threshold for meeting the application requirements of materials in high-energy laser systems. For

instance, previous studies have demonstrated that the laser-induced damage threshold of  $\text{HgBr}_2$  reaches  $0.3 \text{ GW cm}^{-2}$ ,<sup>35</sup> significantly surpassing those of the  $[\text{BS}_3]$ -based compounds, such as  $\text{Ca}_2\text{La}(\text{BS}_3)(\text{SiS}_4)$  ( $34.43 \text{ MW cm}^{-2}$ ),  $\text{LaBS}_3$  ( $49.55 \text{ MW cm}^{-2}$ ) and  $\text{BaB}_2\text{S}_4$  ( $265 \text{ MW cm}^{-2}$ ).<sup>42-44</sup> Although the polarization anisotropy of  $[\text{HgQ}_2]$  is 1.7 to 2 times that of  $[\text{BS}_3]$ , their HOMO-LUMO gap is narrower, which would affect their optical band gap. For the above-mentioned reason, the  $[\text{HgX}_2]$  units are potential building units with comparable performance to  $[\text{BS}_3]$ , suggesting that Hg-based binary halides composed of  $[\text{HgX}_2]$  may serve as possible wide-spectrum birefringent materials. Considering the hydrolysis susceptibility of  $\text{HgCl}_2$  and the narrow band gap exhibited by  $\text{HgI}_2$ , we selected  $\text{HgBr}_2$  as our subject for investigating birefringence properties.

The crystal growth procedure of  $\text{HgBr}_2$  is as follows: a saturated solution of mercury bromide was prepared in a beaker using hot anhydrous ethanol as the solvent under continuous agitation. Subsequently, the clear solution was filtered while still hot into another beaker, and the solvent was very slowly evaporated at room temperature. After one month, a substantial quantity of crystals with two to four millimeter-sized were successfully acquired. The analytical results based on the crystallographic data obtained by single-crystal XRD demonstrate that  $\text{HgBr}_2$  crystal crystallizes in the orthorhombic space group  $Cmc2_1$ , and the cell parameters are:  $a = 4.6215(6) \text{ \AA}$ ,  $b = 6.7794(7) \text{ \AA}$ ,  $c = 12.4277(19) \text{ \AA}$ , and  $Z = 4$ , which are consistent with the results reported in previous literature.<sup>43</sup> The structure of  $\text{HgBr}_2$  is a simple zero-dimensional (0D) arrangement consisting of isolated  $[\text{HgBr}_2]$  units (Fig. 1b), with one crystallographically independent  $\text{Hg}$  atom and two  $\text{Br}$  atoms in each asymmetric unit. The bond lengths of  $\text{Hg-Br}$  range from  $2.427(6) \text{ \AA}$  to  $2.445(7) \text{ \AA}$ , and the angle of  $\angle \text{Br-Hg-Br}$  is  $179.58^\circ$ , both of which align with the corresponding values documented in the literature.<sup>45</sup> In general, the majority of three-dimensional (3D) structural frameworks constructed from metal-centered coordination tetrahedra typically exhibit low anisotropy and small birefringence, whereas lower-dimensional structures tend to display significant anisotropy and consequently induce appreciable birefringence.<sup>46</sup> Therefore, the combination of the 0D structure of  $\text{HgBr}_2$  and the 1D linear building unit  $[\text{HgBr}_2]$  with a large polarizability anisotropy will facilitate the generation of large birefringence.

The optical band gap of  $\text{HgBr}_2$  gained by UV-vis-NIR diffuse-reflectance spectroscopy processed by the Tauc plot method was 3.60 eV (Fig. 2c),<sup>47</sup> which was slightly higher than the band gap reported in previous literature (3.3 eV),<sup>35</sup> larger than those of some commercial IR optical materials such as  $\text{AgGaS}_2$  (2.76 eV),  $\text{AgGaSe}_2$  (1.83 eV),<sup>48</sup>  $\text{ZnSe}$  (2.58 eV) and comparable to that of  $\text{ZnS}$  (3.72 eV).<sup>49,50</sup> The wider optical band gap can effectively mitigate the damage caused by photon absorption to the material, thereby enhancing the laser-induced damage threshold for application in high-energy laser systems. In addition, according to the diffuse reflection spectrum, the UV cutoff edge of  $\text{HgBr}_2$  was about 344 nm. Combined with the IR spectrum, it can be determined that the transparent range of  $\text{HgBr}_2$  was 0.34–22.9  $\mu\text{m}$  (Fig. 2c), covering a broad spectral range from UV to far-IR, surpassing those of all commercially



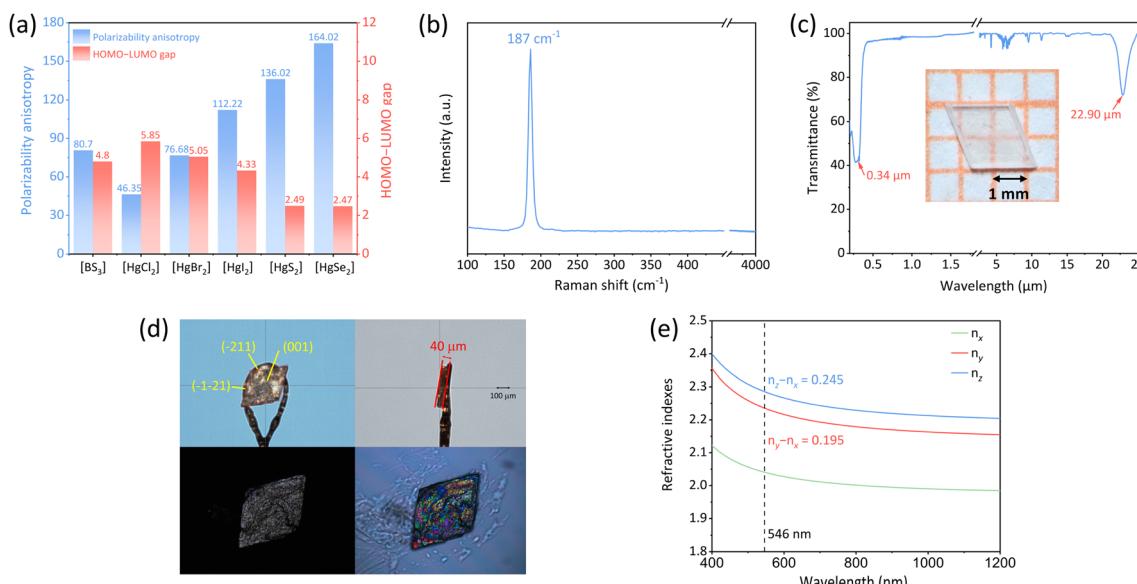


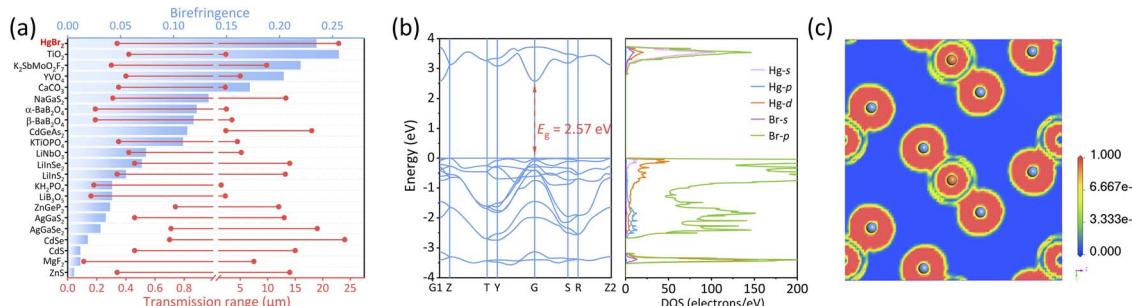
Fig. 2 (a) Polarizability anisotropy and HOMO–LUMO gaps of the  $[\text{BS}_3]$ ,  $[\text{HgCl}_2]$ ,  $[\text{HgBr}_2]$ ,  $[\text{HgI}_2]$ ,  $[\text{HgS}_2]$ , and  $[\text{HgSe}_2]$  units; (b) experimental Raman spectrum of  $\text{HgBr}_2$ ; (c) UV-vis-IR transmittance spectrum collected from the single crystal (inside: the photograph of single crystal  $\text{HgBr}_2$  plate); (d) crystallographic orientation and birefringence measurement of the  $\text{HgBr}_2$  crystal; (e) the calculated birefringence values of  $\text{HgBr}_2$ .

available birefringent materials. The wide transparency range of  $\text{HgBr}_2$  can be attributed to two aspects: the substantial atomic weight of mercury and bromine atoms leads to their low phonon energy thus enabling high transmittance in the far-IR region; additionally, the high electronegativity of bromine leads to a strong bond of valence electrons, consequently yielding a broad optical band gap. In addition, the sharp absorption peak at  $4.2 \mu\text{m}$  in Fig. 2c originates from the asymmetric stretching vibration of the  $\text{C}=\text{O}$  double bond in carbon dioxide molecules; the dense and sharp absorption peaks observed within the range of  $5\text{--}8 \mu\text{m}$  may be attributed to the bending vibrations of hydroxyl groups in gaseous water molecules; the two minor absorption peaks detected at wavelengths of  $9.5 \mu\text{m}$  and  $11.3 \mu\text{m}$  can be ascribed to stretching and bending vibrations associated with  $\text{C}-\text{O}$  single bonds present in ethanol molecules. Given our utilization of ethanol as a solvent for crystal growth purposes, it is possible for weak IR absorption features related to ethanol to emerge within the IR spectrum. The Raman spectrum of  $\text{HgBr}_2$  is shown in Fig. 2b. The peak at  $187 \text{ cm}^{-1}$  has roots in the vibration of the  $\text{Hg}-\text{Br}$  bond,<sup>51</sup> and no other peaks are found in the range from  $4000$  to  $100 \text{ cm}^{-1}$ .

To thoroughly investigate the potential of  $\text{HgBr}_2$  as a birefringent material, crystal  $\text{HgBr}_2$  was measured using a polarizing microscope equipped with a  $546 \text{ nm}$  light source. The crystal of  $\text{HgBr}_2$  chosen for the birefringence test is depicted in Fig. 2d. The crystallographic plane of the single crystal for birefringence measurement was confirmed as (001) through orientation using single-crystal XRD. Referring to the Michel-Levy Birefringence Chart, the optical path difference of the measured crystal was  $9411.707 \text{ nm}$ , and its thickness was  $0.04 \text{ mm}$ , and thus a calculated value of  $0.235$  can be assigned to the refractive index difference of this crystal under  $546 \text{ nm}$

illumination, which surpassed that of commercially available birefringent material  $\text{MgF}_2$  ( $0.012$ )<sup>16</sup> and even those of recently reported binary halides  $\text{SbCl}_3$  ( $0.172$ ) and  $\alpha\text{-SnF}_2$  ( $0.177$ ) containing stereochemical activity lone pairs.<sup>52</sup> Since  $\text{HgBr}_2$  crystallized in the orthorhombic system and belonged to a biaxial crystal, there were three refractive indices ( $n_x$ ,  $n_y$ , and  $n_z$ ) along the  $x$ -,  $y$ -, and  $z$ -axes, respectively. The relationship between these refractive indices was given as  $n_z > n_y > n_x$ . The maximum birefringence of  $\text{HgBr}_2$  occurred at the (010) crystallographic plane, which corresponded to the difference between  $n_z$  and  $n_x$ . However, due to the intrinsic growth habit of  $\text{HgBr}_2$ , it was challenging to expose the (010) plane effectively. Despite attempts to mechanically cut the crystal to obtain an exposed (010) plane, we have been unsuccessful in producing a wafer that met the testing conditions of a polarizing microscope. Therefore, considering that the measured crystal plane of  $\text{HgBr}_2$  was not parallel to the optical axis plane, the birefringence obtained at the (001) crystal plane should be less than or equal to  $n_z - n_x$  and close to  $n_y - n_x$ . As presented in Fig. 2e, the maximum calculated refractive index difference ( $n_z - n_x$ ) of  $\text{HgBr}_2$  was  $0.245$  at  $546 \text{ nm}$  and the value of  $n_y - n_x$  was  $0.195$ , which exhibited good agreement with the experimental birefringence measurement and our conjecture. Despite challenges associated with exposing specific crystallographic planes in the  $\text{HgBr}_2$  crystal for accurate measurement purposes, our findings demonstrate that there is still agreement between theoretical calculations and experimental results when it comes to understanding its birefringent properties on other crystal planes, which has not received enough attention in previous similar studies.

We have conducted a statistical analysis for classical birefringent materials ( $\text{MgF}_2$ ,  $\alpha\text{-BaB}_2\text{O}_4$ ,  $\text{YVO}_4$ ,  $\text{TiO}_2$ ,  $\text{LiNbO}_3$ ,  $\text{CaCO}_3$ ),<sup>16–21</sup> recently reported birefringent materials with large



**Fig. 3** (a) Comparison of  $\text{HgBr}_2$  with typical optical crystals in terms of birefringence and transmission range; (b) the calculated band structure (left) and density of states (DOS, right) of  $\text{HgBr}_2$ ; (c) 2D ELF diagram through the  $[\text{Br}-\text{Hg}-\text{Br}]$  motif for  $\text{HgBr}_2$ .

crystals ( $K_2SbMoO_2F_7$ ,  $NaGaS_2$ ),<sup>46,53</sup> common commercial nonlinear optical materials ( $LiB_3O_5$ ,  $\beta\text{-BaB}_2O_4$ ,  $KH_2PO_4$ ,  $KTiOPO_4$ ,  $AgGaS_2$ ,  $AgGaSe_2$ ,  $ZnGeP_2$ ,  $CdGeAs_2$ ),<sup>48</sup> and some chalcogenide optical materials ( $ZnS$ ,  $CdS$ ,  $CdSe$ )<sup>54-56</sup> to compare them with  $HgBr_2$  in terms of transparent range and birefringence. Note that the transparent spectra of these materials were tested based on large crystals, ensuring high reliability. Additionally, most of the refractive index tests for these materials were performed using an  $\sim 550$  nm light source, except for narrow band gap materials including  $AgGaSe_2$  (@2000 nm),  $CdSe$  (@2000 nm),  $ZnGeP_2$  (@2000 nm) and  $CdGeAs_2$  (@3000 nm). From Fig. 3a, it is evident that  $HgBr_2$  ranked among the top in both the spectral transparent range and birefringence among these optical materials; specifically,  $HgBr_2$  possessed a transparent range that exceeds those of most birefringent materials, and its birefringent performance is second only to  $TiO_2$  among commercial birefringent materials, which demonstrated that  $HgBr_2$  has great potential as a wide-spectrum birefringent material.

In order to better understand the relationship between the structure and properties, the electronic structure of  $\text{HgBr}_2$  has been calculated based on the first-principles method. As depicted in Fig. 3b (left),  $\text{HgBr}_2$  was a direct bandgap semiconductor with a calculated band gap of 2.57 eV, which slightly deviates from the experimental optical band gap (3.60 eV) due to the inherent limitation of the GGA-PBE functional caused by the discontinuity of exchange correlation energy, resulting in an underestimation of the band gap.<sup>57</sup> However, this value is consistent with the previous calculation by Kanchana *et al.* without using the TB-mBJ functional (2.24 eV).<sup>58</sup> The DOS of  $\text{HgBr}_2$  is shown in Fig. 3b (right). The top of the valence band (VB) and the bottom of the conduction band (CB) were mainly governed by  $\text{Hg-6s}$ ,  $\text{Hg-5d}$ , and  $\text{Br-4p}$  orbitals, indicating that the optical properties of  $\text{HgBr}_2$  were predominantly influenced by  $[\text{HgBr}_2]$  units. The electron localization function (ELF) maps projected on crystal planes parallel to  $[\text{HgBr}_2]$  motifs were further subjected to analysis. As illustrated in Fig. 3c, the electron cloud around the  $[\text{HgBr}_2]$  motif exhibited a dumbbell-like shape and demonstrates large optical anisotropy, which was the primary cause of the significant birefringence in  $\text{HgBr}_2$ . The above results further validated the rationality of the experimental findings.

## Conclusions

In summary, we have investigated  $\text{HgBr}_2$  as a wide-spectrum birefringent material for the first time and successfully obtained a large number of millimeter-scale single crystals through simple and mild growth methods. The findings revealed that  $\text{HgBr}_2$  possessed a broad band gap (3.60 eV) and transparent range (0.34–22.9  $\mu\text{m}$ ), and notably, it displayed a large birefringence that is 19.6 times that of the commercial halide  $\text{MgF}_2$ , thereby demonstrating  $\text{HgBr}_2$  as a promising wide-spectrum birefringent material. Furthermore, we substantiated for the first time *via* theoretical calculations that the Hg-based linear unit possessed a significant polarizability anisotropy, qualifying it as a potential candidate for constructing superior birefringent materials. This work will provide a novel idea for the design and exploration of wide-spectrum birefringent materials in the future.

## Data availability

Data available on request from the authors.

## Author contributions

Ming-Shu Zhang: conceptualization, data curation, visualization, writing – original draft; Wen-Dong Yao: theoretical calculations; Shao-Min Pei: data curation, theoretical calculations; Bin-Wen Liu: writing – review & editing; Xiao-Ming Jiang: software; Guo-Cong Guo: supervision.

## Conflicts of interest

There are no conflicts to declare.

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