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# Improving the gas sorption capacity in lantern-type metal–organic polyhedra by a scrambled cage method†

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The synthesis of multivariate metal–organic frameworks (MOFs) is a well-known method for increasing the complexity of porous frameworks. In these materials, the structural differences of the ligands used in the synthesis are sufficiently subtle that they can each occupy the same site in the framework. However, multivariate or ligand scrambling approaches are rarely used in the synthesis of porous metal–organic polyhedra (MOPs) – the molecular equivalent of MOFs – despite the potential to retain a unique intrinsic pore from the individual cage while varying the extrinsic porosity of the material. Herein we directly synthesise scrambled cages across two families of lantern-type MOPs and find contrasting effects on their gas sorption properties. In one family, the scrambling approach sees a gradual increase in the BET surface area with the maximum and minimum uptakes associated with the two pure homoleptic cages. In the other, the scrambled materials display improved surface areas with respect to both of the original, homoleptic cages. Through analysis of the gas sorption isotherms, we attribute this effect to the balance of micro- and mesoporosity within the materials, which varies as a result of the scrambling approach. The gas uptake of the materials presented here underscores the tunability of cages that springs from their combination of intrinsic, extrinsic, micro- and meso-porosities.

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## Introduction

Metal–organic polyhedra (MOPs) or metal–organic cages (MOCs) are assembled through the coordination of organic ligands to metal nodes giving discrete molecules that contain a defined cavity.<sup>1–5</sup> The shape of this cavity arises from the combination of the bonding preferences of the metal node with the geometry and denticity of the ligand(s) used in the synthesis.<sup>6,7</sup> Homoleptic cages contain one type of ligand, and examples of this approach include the use of 3,3'-((1,3-phenylene)bis(ethyne-2,1-diyl))dibenzoic acid or 1,3-bis(pyridin-3-ylethynyl)benzene derivatives to synthesise lantern-type cages,<sup>8–13</sup> carbazole derivatives to access octahedral

cages;<sup>14–16</sup> and isophthalic acid or 1,3-di(pyridine-4-yl)benzene derivatives to synthesise cuboctahedral cages.<sup>17–23</sup> Alternatively, two or more types of ligand can be used, giving heteroleptic cages, increasing the complexity of the assembly as now the geometry of the cage will be determined by contributions from the bonding angles as well as the relative lengths of the ligands. In the study of self-assembly of metal–organic cages in solution, this approach is used as a way to create more complex micro-environments within the cavity of the cage itself.<sup>24–29</sup> When MOPs have been investigated for gas sorption in the solid-state, heteroleptic approaches have largely involved the use of two ligands with well-defined, independent structural roles,<sup>30–32</sup> such as the use of calixarenes to form bowl-type arrangements, which are then connected through isophthalates or other carboxylate-based ligands to complete the discrete cage.<sup>33–36</sup>

The above routes focus on the formation of the internal pore environment. However, the porosity of MOPs in the solid state also depends on the surface functionalisation of the cage,<sup>37–39</sup> as this affects packing and hence, the extrinsic porosity. In this context, heteroleptic MOPs where the ligands used have the same backbone, but different external functionalisation, are less well explored as a means of tuning porosity in the cages. An early example of this type of MOP is [Cu<sub>24</sub>(OH-bdc)<sub>12</sub>(bdc)<sub>12</sub>], which was obtained *via* solution-phase ligand exchange on the cuboctahedral MOP [Cu<sub>24</sub>(OH-bdc)<sub>24</sub>], where OH-bdc and bdc

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represent 5-hydroxyisophthalate and isophthalate, respectively.<sup>40</sup> In this case the effect of the ligand exchange on the gas sorption properties of the material was not reported. These processes require the MOP to present a high degree of solubility, which is the case for  $[\text{Cu}_{24}(\text{OH-bdc})_{24}]$ ,<sup>22,41</sup> and for the rate of ligand exchange to be favourable.<sup>42,43</sup> Recent work from von Baeckmann and co-workers described a stepwise strategy based on protection/deprotection strategies available to more robust  $[\text{Rh}_2]$  paddlewheels.<sup>44</sup> If the MOPs are soluble, but no ligand exchange takes place, then pure homoleptic cages can also be blended to obtain alloys.<sup>45</sup> When direct synthesis is employed with ligands of the same length, then the approach is similar to that found in multivariate MOFs,<sup>46</sup> or analogous to the dynamic covalent scrambling that is possible for porous organic cages (POCs).<sup>47–49</sup> Previous work in this area for porous MOPs is limited to a few examples.<sup>50</sup> Lerma-Berlanga and co-workers used this method to develop a family of MOPs based on mononuclear Ti(IV) nodes, where the different hydrogen bond donor/acceptor capabilities of the MeO- and -NH<sub>2</sub> functionalised ligands were found to affect the degree of incorporation of the ligands in the final cages.<sup>51</sup> Antonio and co-workers<sup>52</sup> and Nam and co-workers<sup>53</sup> found that the surface areas of scrambled cages lay between those of pure homoleptic materials for  $[\text{Mo}_{24}(\text{bdc})_{24}]$ -type cuboctahedra and tetrahedra based on  $[(\text{Cp}_3\text{Zr})_3]$  nodes, respectively.

Recently, we reported the gas sorption properties of a family of homoleptic lantern-type cages with the general formula  $[\text{Cu}_4(\text{RL})_4]$ , where  $\text{RL}^{2-}$  was a series of derivatives of the ligand  $\text{LH}_2$  (Fig. 1).<sup>54</sup> The homoleptic lantern  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$  was found to display the highest Brunauer–Emmett–Teller (BET) surface area for lantern cages, as derived from the N<sub>2</sub> sorption isotherm. Herein, we target the formation of scrambled cages by using mixtures of the ligands  $\text{LH}_2$ ,  $\text{CH}_3\text{LH}_2$ , and  $\text{MeOLH}_2$  (Fig. 1) to investigate two families of MOPs:  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$

and  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ , for  $x = 0, 1, 2, 3$ , and 4. Using this approach, we demonstrate that it can be used to tune the gas sorption properties in MOPs and, notably, the scrambled material  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$  displays the highest BET surface area observed to date for lantern-type complexes, with improved gas uptake compared to both of the parent homoleptic materials.

## Results and discussion

### Synthesis of the cage families $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ and $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$

Although solution-based ligand exchange methods have been used to obtain heteroleptic porous coordination cages based on Cu(II),<sup>40,55</sup> the poor solubility of the as-formed homoleptic cages  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ -DMA and  $[\text{Cu}_4(\text{MeOL})_4]$ -DMA precluded that approach here. Instead, a given ratio of the ligands  $\text{CH}_3\text{LH}_2$  and  $\text{LH}_2$ , or  $\text{CH}_3\text{LH}_2$  and  $\text{MeOLH}_2$ , was reacted with  $\text{Cu}(\text{OAc})_2 \cdot \text{H}_2\text{O}$  in dimethylacetamide (DMA) overnight at 80 °C, yielding single crystals of the complexes. Attempts to synthesise the series  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{L})_x]$  led to phases with poor reproducibility, and were not studied further. In addition, we synthesised a new polymorph of the cage  $[\text{Cu}_4(\text{L})_4]$ -DMA, using  $\text{Cu}(\text{OAc})_2 \cdot \text{H}_2\text{O}$  as the metal source (Table S1†) – this cage had previously been reported in the space group  $P2_1/c$  when synthesised from  $\text{CuCl}_2$ .<sup>56</sup> For the cages  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ -DMA ( $x = 1, 2, 3$ ), the single crystal X-ray diffraction data were of sufficient quality to extract the unit cell parameters (Table S2†) and show the connectivity of the core of the lantern (Fig. S1†), but were too highly disordered to accurately model the positions of the methoxy- and methyl substituents of the cages, in common with other attempts to obtain scrambled metal–organic polyhedra.<sup>52</sup> These scrambled cages crystallise in the monoclinic space group  $P2_1/n$ , as does the homoleptic cage  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ -DMA, but differing from the cage  $[\text{Cu}_4(\text{MeOL})_4]$ -DMA ( $C2/c$ ). All of the

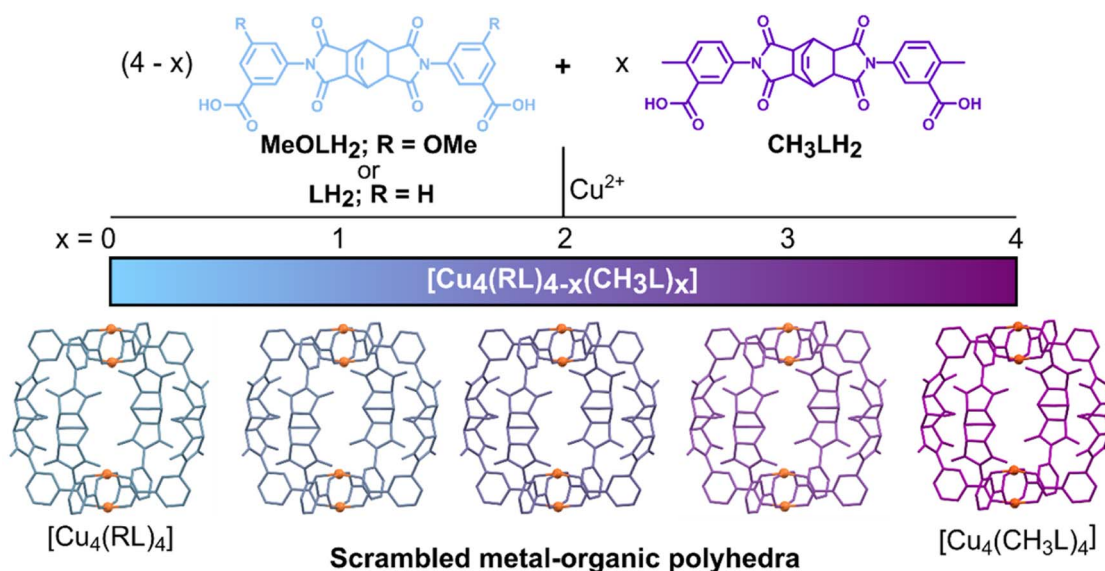


Fig. 1 Schematic of the approach used to obtain the scrambled metal–organic polyhedra. The relevant ratios of either  $\text{MeOLH}_2$  or  $\text{LH}_2$  were mixed with  $\text{CH}_3\text{LH}_2$  and reacted with  $\text{Cu}(\text{II})$  salts to obtain scrambled materials. The crystal structure of the material  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$ -DMA was used to prepare the representations of the scrambled MOPs, omitting the functional groups on the cages.



cages are neutral, formed by two  $[\text{Cu}_2]$  paddlewheel nodes linked through four deprotonated organic linkers. As the crystallographic data were too disordered to resolve the positions of the methyl and methoxy-groups, IR spectroscopy was used to track the scrambling of the cages, illustrated by the decrease in the intensity of the stretch at  $770\text{ cm}^{-1}$  as the content of the ligand  $\text{MeOL}^{2-}$  decreases (Fig. 2(a) and S5<sup>†</sup>). Comparison of the powder X-ray diffraction patterns show that the scrambled cages display similar packing (Fig. 2(b) and S6<sup>†</sup>). To prove that the scrambled cages were not a mixture of crystallites, a physical mixture of the homoleptic cages  $[\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$  and  $[\text{Cu}_4(\text{MeOL})_4]\text{-DMA}$  was prepared for comparison with the scrambled sample  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ . The physical mixture was

formed by grinding together separately synthesised samples of the homoleptic cages until formation of a homogenous powder  $[\text{Cu}_4(\text{MeOL})_4]\text{-DMA} + [\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$ , with this powder showing features from the separate diffractograms corresponding to the unique cages (Fig. S8–S9<sup>†</sup>), distinct to the diffractogram for  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ . From  $^1\text{H-NMR}$  digestion experiments, the connectivity observed in the SXRD data, and thermogravimetric analysis (Fig. S10–S16<sup>†</sup>), the bulk, averaged composition of the scrambled cages is proposed to be  $[\text{Cu}_4(\text{MeOL})_3(\text{CH}_3\text{L})(\text{DMA})_2(\text{H}_2\text{O})_2] \cdot 7\text{DMA} \cdot 2\text{H}_2\text{O}$ ,  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2(\text{DMA})_2(\text{H}_2\text{O})_2] \cdot 6\text{DMA} \cdot 2\text{H}_2\text{O}$ , and  $[\text{Cu}_4(\text{MeOL})(\text{CH}_3\text{L})_3(\text{DMA})_2(\text{H}_2\text{O})_2] \cdot 7\text{DMA} \cdot \text{H}_2\text{O}$ , for the cages  $[\text{Cu}_4(\text{MeOL})_3(\text{CH}_3\text{L})]\text{-DMA}$ ,  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ , and  $[\text{Cu}_4(\text{MeOL})(\text{CH}_3\text{L})_3]\text{-DMA}$ , respectively.

The second family of scrambled cages based on varying ratios of  $\text{CH}_3\text{LH}_2$  and  $\text{LH}_2$  were suitable for single-crystal X-ray analysis. Across the series  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]\text{-DMA}$  ( $x = 0, 1, 2, 3,$  and  $4$ ), all of the cages crystallise in the monoclinic space group  $P2_1/n$ . DMA molecules coordinate to the outer sites of the paddlewheel while water molecules are coordinated in the inner site of the paddlewheel (Fig. S18 and S19<sup>†</sup>). The asymmetric unit of each member of the family consists of one half of a cage, with two crystallographically independent positions for the ligands (Fig. 3(a)). Therefore, rather than occupying a unique position within the MOP, yielding an ordered heteroleptic cage of unique composition, the ligand  $\text{CH}_3\text{L}^{2-}$  is split over both possible sites. It was possible to refine these occupancies, as shown in Fig. 3 for the cage  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ , where the occupancy of one site by  $\text{CH}_3\text{L}^{2-}$  is 64% and in the other it is 53%. Because of the previously mentioned poor solubility of the cages, solution-phase analysis techniques such as mass spectrometry were not possible to evaluate the relative populations of the distinct possible cage compositions. Based on the occupancies determined from the crystallographic data, the most probable composition of a cage in the material  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]\text{-DMA}$  is  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$  (35.7%), while the next most probable composition is  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]$  (33.3%). Similarly, for  $[\text{Cu}_4(\text{L})_3(\text{CH}_3\text{L})]\text{-DMA}$  it is  $[\text{Cu}_4(\text{L})_3(\text{CH}_3\text{L})]$  (42.6%), and for  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]\text{-DMA}$  it is  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]$  (43.8%). For all of the scrambled MOPs there is a significant presence of the other possible compositions in the crystal structure, but the statistically most likely composition matches the composition that was targeted (Fig. S20<sup>†</sup>). The average composition of the cages as determined by NMR digestion experiments is close to that determined from the single crystal diffraction data (Fig. S32–S34<sup>†</sup>), with the discrepancies attributed to the uncertainty arising from the disorder in the structure. As the content of the ligand  $\text{CH}_3\text{L}^{2-}$  increases within the material, a smooth increase in the length of the  $c$ -axis of the  $P2_1/n$  unit cell is observed, from  $24.9013(2)$  to  $25.7952(1)\text{ \AA}$  for the homoleptic cages  $[\text{Cu}_4(\text{L})_4]\text{-DMA}$  and  $[\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$  at either end of the series, respectively. This behaviour fits with Vegard's law, as has been observed previously for alloys of porous organic cages, and in keeping with the solid solution behaviour of the samples (Fig. 3(b)).<sup>57</sup> This trend is due to the crystal packing of the molecules, in which the methyl groups of the ligand  $\text{CH}_3\text{L}^{2-}$  point approximately along the  $c$ -axis, causing the increase in the

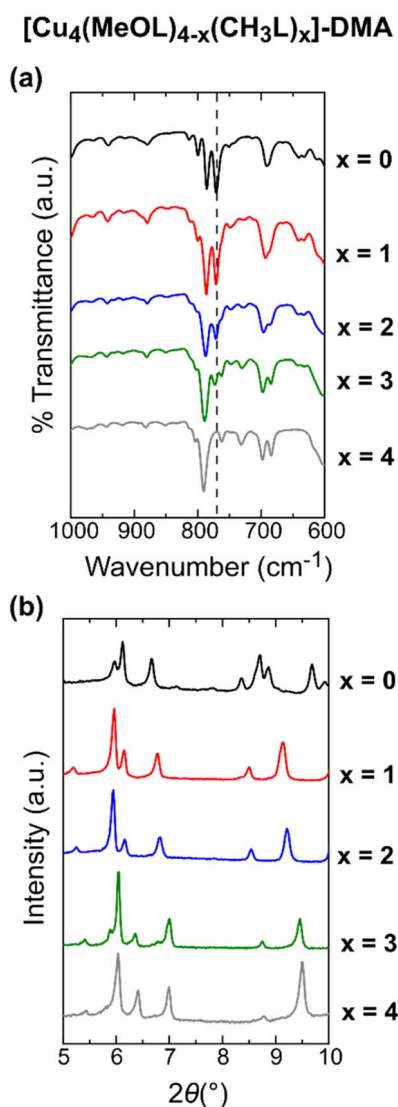
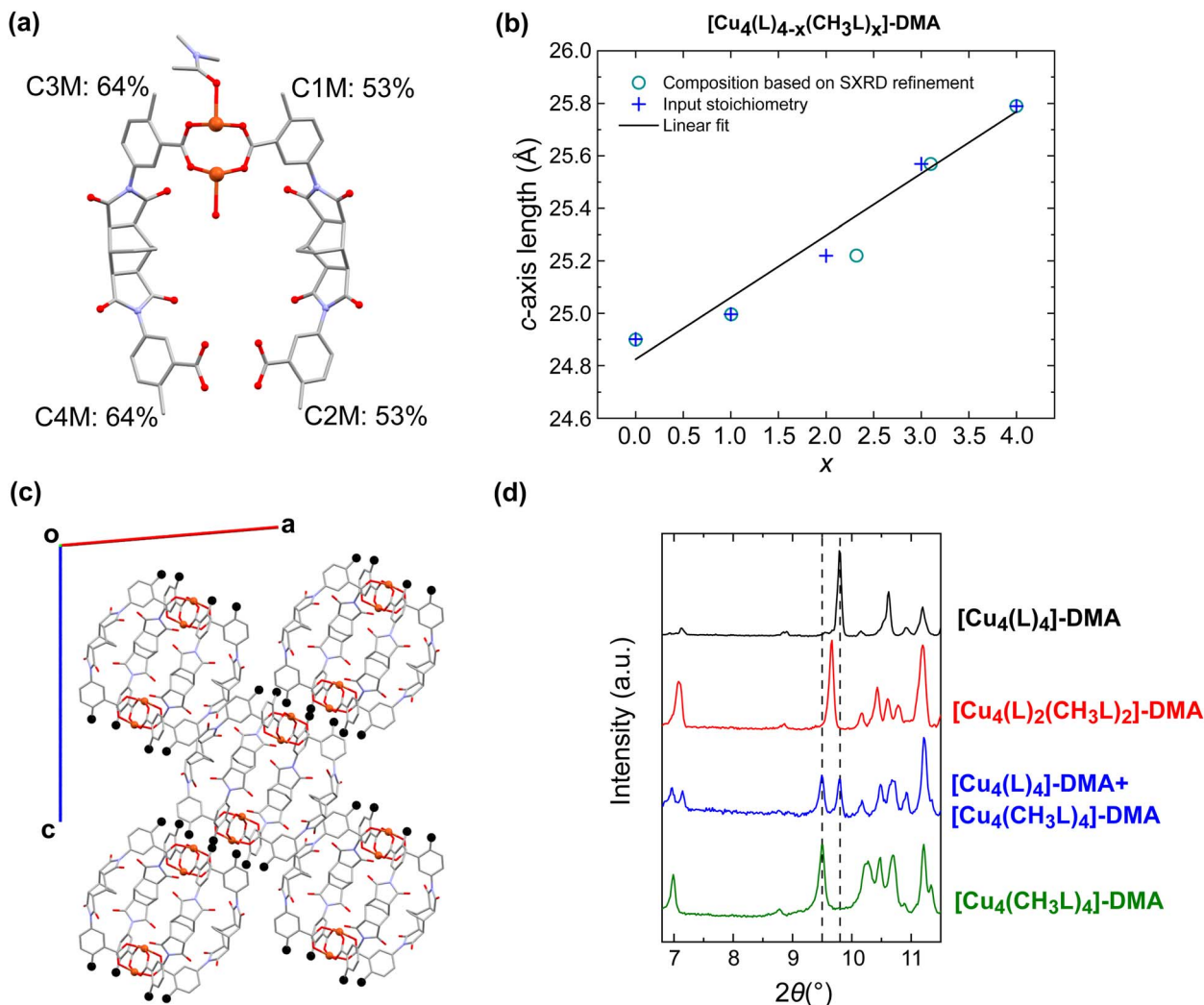


Fig. 2 (a) IR spectra in the region  $1000\text{--}600\text{ cm}^{-1}$  for the family of MOPs  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]\text{-DMA}$  ( $x = 0, 1, 2, 3, 4$ ). The dashed line highlights the decreasing intensity of a stretch at  $770\text{ cm}^{-1}$  as  $x$  increases. (b) PXRD data for  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]\text{-DMA}$  ( $x = 0, 1, 2, 3, 4$ ), showing the isostructural nature of the crystalline powders of the scrambled cages, which all crystallise in  $P2_1/n$ , while the homoleptic cage obtained for  $x = 0$  crystallises in  $C2/c$ . The data for  $x = 0$  and  $4$  were previously published.<sup>54</sup>





**Fig. 3** (a) View of the asymmetric unit for the isostructural cages  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]\text{-DMA}$ . The percentages shown represent the occupancy of the methyl functional groups in the cage for the case  $x = 2$ . Hydrogen atoms and all non-coordinated solvent molecules have been omitted for clarity. (b) Plot of the dependence of the length of the *c*-axis of the unit cell on the content of  $\text{CH}_3\text{L}^{2-}$  in the cages. The composition of the cages derived from refinement of the SXRD data is shown together with the input stoichiometry of the ligand, and the line represents a straight line fit. (c) View along the *b*-axis of the cage  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ , with the methyl groups shown as black spheres to emphasise that they point along the *c*-axis. (d) Comparison of the powder X-ray diffraction data for the homoleptic cages, the scrambled cage  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ , and the physical mixture  $[\text{Cu}_4(\text{L})_4]\text{-DMA} + [\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$ . The dashed lines correspond to  $2\theta = 9.50$  and  $9.80^\circ$ .

cell dimension as the occupancy increases (Fig. 3(c)). To confirm the solid-solution like properties of the as-synthesised scrambled cages, again a physical mixture of the cages  $[\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$  and  $[\text{Cu}_4(\text{L})_4]\text{-DMA}$  was made, with the resulting sample labelled as  $[\text{Cu}_4(\text{L})_4]\text{-DMA} + [\text{Cu}_4(\text{CH}_3\text{L})_4]\text{-DMA}$ . Powder X-ray diffraction of the physical mixture displays two, well defined peaks arising from the  $[1\ 1\ -2]$  planes of each homoleptic MOP, while the diffractogram obtained for  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]\text{-DMA}$ , shows one unique peak at  $2\theta = 9.65^\circ$ , confirming the unique phases obtained through the direct synthesis of the scrambled cages (Fig. 3(d) and S37†).

### Solvent-exchanged phases

The scrambled cages were soaked in MeOH to exchange out the DMA molecules prior to activation and gas sorption

measurements. All samples were characterised *via* IR spectroscopy, PXRD, TGA, and  $^1\text{H-NMR}$  spectroscopy of digested samples, with the data presented in the ESI† IR spectra of the samples (Fig. S39 and S40†) show that the characteristic stretching vibration highlighted in Fig. 2(a) is still present in all of the corresponding cages with its relative intensity unchanged, suggesting that the composition of the cages is unaltered after solvent exchange. The  $^1\text{H-NMR}$  spectra (Fig. S41–S47†) of acid-digested samples confirm that the composition of the cages in terms of the ligand ratios remains the same, with successful exchange of DMA for MeOH taking place, while the TGA data show that the overall thermal stability of the cages is retained. The PXRD data for  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]\text{-MeOH}$  show that as  $x$  increases, the crystallinity of the bulk powder also increases (Fig. 4). A similar, although less



pronounced, effect is also observed for the family  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]\text{-MeOH}$  (Fig. S48†), although the overall effect of the methoxy substituent appears to be a reduction in the crystallinity of this family of cages compared to the family  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ .

### Gas sorption properties

To assess the effect of scrambling on the gas sorption properties of the cages, the uptake of  $\text{N}_2$  at 77 K and  $\text{CO}_2$  at 195 K was measured. We have also measured the previously unreported gas sorption properties for the non-functionalised, homoleptic cage  $[\text{Cu}_4(\text{L})_4]$ . Previously, we reported the gas sorption isotherms for the homoleptic cages  $[\text{Cu}_4(\text{MeOL})_4]$  and  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ , and those data are included here for comparison. Fig. 5 displays the  $\text{N}_2$  adsorption isotherms for both families of cages measured at 77 K (the desorption data, and isotherms for  $\text{CO}_2$  are provided in the ESI, Fig. S54–S64†). While the homoleptic cage  $[\text{Cu}_4(\text{MeOL})_4]$  does not present  $\text{N}_2$  uptake at 77 K,  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$  displays the steep uptake associated with microporous materials, and BETSI was used to determine a BET surface area of  $380 \text{ m}^2 \text{ g}^{-1}$ .<sup>58</sup> Here, all of the scrambled materials present Type H4 isotherms.<sup>59</sup> As  $x$  increases for  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ , the uptake of  $\text{N}_2$  is shown to improve significantly for  $x \geq 3$ . For  $[\text{Cu}_4(\text{MeOL})(\text{CH}_3\text{L})_3]$ , the uptake at  $P/P_0 \approx 0.9$  is of  $85.8 \text{ cm}^3 \text{ g}^{-1}$ , compared to  $5.8 \text{ cm}^3 \text{ g}^{-1}$  for  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]$ , and  $117.4 \text{ cm}^3 \text{ g}^{-1}$  for  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ . This is reflected by an increase in the BET surface areas as  $x$  increases (Fig. 6), reaching a maximum for the homoleptic material  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ . This trend, in which mixed cages show gas uptake that lies between that found for pure homoleptic cages has been observed for tetrahedra based on  $[\text{Cp}_3\text{Zr}_3]$  nodes and cuboctahedra containing  $[\text{Mo}_2]$  paddlewheels linked with functionalised isophthalate ligands.<sup>52,53</sup> In addition, we measured the  $\text{N}_2$  uptake for the physical mixture  $[\text{Cu}_4(\text{MeOL})_4] + [\text{Cu}_4(\text{CH}_3\text{L})_4]$ , and found that it displayed superior gas uptake to the scrambled material  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]$  (Fig. S60†). We propose that

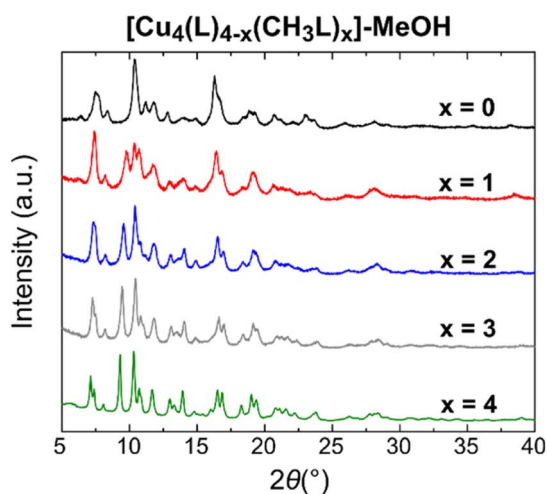


Fig. 4 PXRD data showing the increase in crystallinity of the MeOH-exchanged cages  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]\text{-MeOH}$  as the content of the  $\text{CH}_3\text{L}^{2-}$  ligand increases.

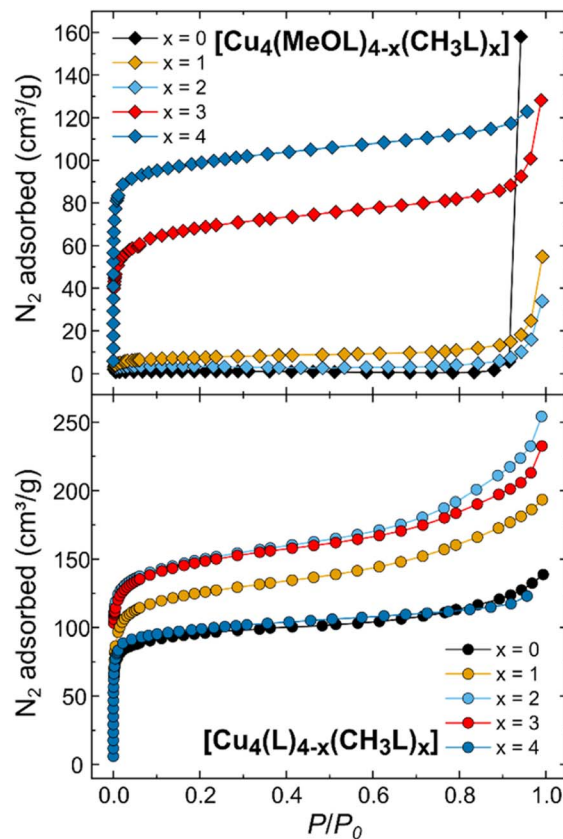


Fig. 5 (Top)  $\text{N}_2$  uptake at 77 K for the family  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ . The data for  $x = 0$  and 4 were previously published, and are provided here for comparison. (Bottom)  $\text{N}_2$  uptake measured at 77 K for the family  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ . The data for  $x = 4$  were previously published and are provided here for comparison.<sup>54</sup> For clarity, only the adsorption branches are shown, and desorption data are given in the ESI.†

this is due to the presence in the physical mixture of the pure homoleptic cage  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ , which may not be present to a large extent in the scrambled material  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]$ . The uptake of  $\text{CO}_2$  at 195 K, of all of the members of the family  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ , is found to be Type I, with a range from  $109.9 \text{ cm}^3 \text{ g}^{-1}$  to  $127.7 \text{ cm}^3 \text{ g}^{-1}$  at  $P/P_0 \approx 0.9$  for  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$  and  $[\text{Cu}_4(\text{MeOL})_3(\text{CH}_3\text{L})]$ , respectively. In this case, the scrambled materials do not greatly improve the  $\text{CO}_2$  sorption capacity with respect to their homoleptic analogues (Fig. S64†). Post-sorption analysis of the scrambled cages  $[\text{Cu}_4(\text{MeOL})_3(\text{CH}_3\text{L})]$ ,  $[\text{Cu}_4(\text{MeOL})_2(\text{CH}_3\text{L})_2]$ , and  $[\text{Cu}_4(\text{MeOL})(\text{CH}_3\text{L})_3]$  using IR spectroscopy and PXRD suggest that the samples retain the same packing as their MeOH solvates, indicating that the samples do not decompose after activation (Fig. S65–S70†). The  $^1\text{H-NMR}$  spectra of digested samples show that the scrambled cage  $[\text{Cu}_4(\text{MeOL})_3(\text{CH}_3\text{L})]$  retains 0.06 MeOH/cage (Fig. S71†), which we attribute to MeOH coordinated to the paddlewheel. Previously, we had observed that subsequent to activation the homoleptic cage  $[\text{Cu}_4(\text{MeOL})_4]$  also retained MeOH, but to a greater degree, with a solvent content of 0.4 molecules of MeOH per molecule of cage.

Next, we studied the effect of scrambling on the gas sorption properties for the family  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ . In this case both



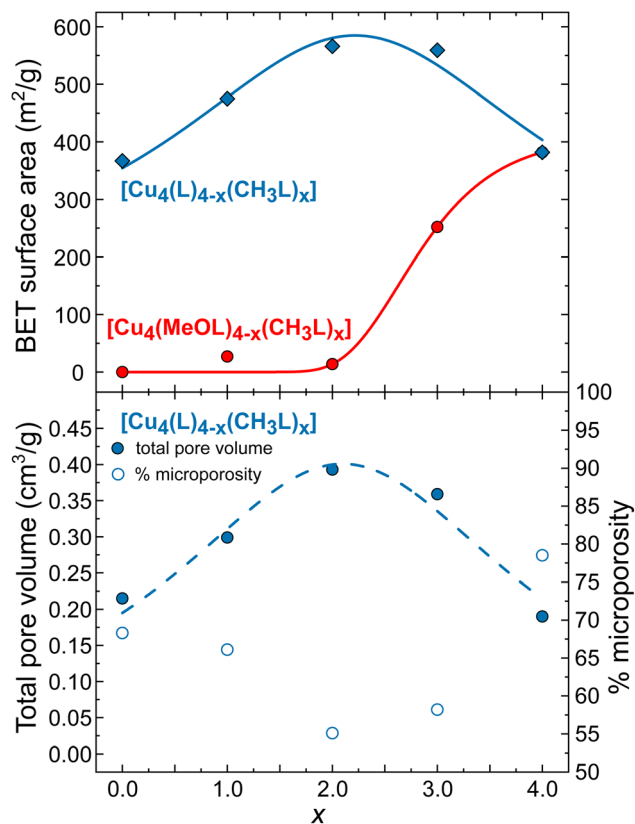


Fig. 6 (Top) Calculated BET surface areas for the families of cages  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$  (filled red symbols) and  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$  (filled blue symbols). The red line and blue lines are guides for the eye derived from sigmoidal and Lorentzian fits, respectively. (Bottom) Plot of the total pore volume (filled symbols) and % microporosity (empty symbols) for the cages  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ . The dashed line is a guide for the eye based on a Lorentzian fit of the data.

homoleptic cages show  $\text{N}_2$  uptake, although the sorption properties of the cage  $[\text{Cu}_4(\text{L})_4]$  had not previously been reported. This homoleptic cage presents a Type I isotherm with rapid uptake of  $\text{N}_2$  at 77 K, reaching  $88.8 \text{ cm}^3 \text{ g}^{-1}$  at  $P/P_0 \approx 0.06$ . At  $P/P_0 \approx 0.92$ , the uptake was of  $123.7 \text{ cm}^3 \text{ g}^{-1}$ . The BET surface area calculated for  $[\text{Cu}_4(\text{L})_4]$  is  $367 \text{ m}^2 \text{ g}^{-1}$  (Fig. S74 and S75†), compared to the previously reported BET surface area of  $382 \text{ m}^2 \text{ g}^{-1}$  for  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ . For the scrambled cages  $[\text{Cu}_4(\text{L})_3(\text{CH}_3\text{L})]$ ,  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$ , and  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]$ , all show increased uptake of  $\text{N}_2$  in the low pressure region  $P/P_0 < 0.1$  when compared to the homoleptic cages. Similarly, all show higher uptake at  $P/P_0 \approx 0.9$ , of 172.6, 217.2, and  $201 \text{ cm}^3 \text{ g}^{-1}$  for  $[\text{Cu}_4(\text{L})_3(\text{CH}_3\text{L})]$ ,  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$ , and  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]$ , respectively. Comparable increases in the adsorption of  $\text{CO}_2$  at 195 K are also found for the scrambled materials. This behaviour is reflected in higher BET surface areas for all of the scrambled cages when compared to the homoleptic materials, of 475, 566, and  $559 \text{ m}^2 \text{ g}^{-1}$  for  $[\text{Cu}_4(\text{L})_3(\text{CH}_3\text{L})]$ ,  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$ , and  $[\text{Cu}_4(\text{L})(\text{CH}_3\text{L})_3]$ , respectively, as determined from the  $\text{N}_2$  adsorption isotherms (Fig. 6 and S76–S81†). This trend is rare in metal-organic cages, and we could only identify one other example where scrambling the cages in this way led to higher surface areas than for the respective homoleptic cages, reported

by Lerma-Berlanga and co-workers.<sup>51</sup> In contrast to the family  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$ , post-sorption analysis of the cages  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$  ( $x = 0, 1, 2,$  and  $3$ ) showed that all of the cages were fully activated, with no remaining MeOH in the  $^1\text{H-NMR}$  spectra collected subsequent to digestion (Fig. S87–S90†). The remaining post-sorption analysis of the samples *via* IR spectroscopy and PXRD (Fig. S91–S98†) confirm that the materials retain the packing of the MeOH solvated phases upon completion of gas sorption measurements.

To understand the origin of the effect of scrambling on the gas sorption properties of both families of cages, we undertook Dubinin–Radushkevich analysis as an approximation to evaluate the relative contributions from micro- and mesoporosity in the cages. This analysis can be used to calculate the micropore volume within a sample, which is derived from the intercept of the Dubinin–Radushkevich plot with the y-axis,  $\log(V)$ , where  $V$  is the adsorbed volume of  $\text{N}_2$  (plots are provided alongside the corresponding  $\text{N}_2$  isotherm in the ESI†).<sup>60,61</sup> This was then compared with the total pore volumes of the solids, which were obtained from the adsorbed volume of  $\text{N}_2$  observed at higher values of  $P/P_0$ . Fig. 6 presents this analysis for the family of cages  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ . The scrambled cages display higher total pore volumes than the respective homoleptic materials. At the same time, the relative contribution to this volume from microporosity decreases: for the scrambled material  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$  this is approximately 55%, compared to 78% for the cage  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ . Therefore, we suggest that the most important effect of the scrambling process is the creation of mesopores, arising from inefficient packing of the cages.

## Conclusions

In this paper, we have used scrambled cage approaches to tune the gas sorption properties of two families of lantern-type MOPs. While we were able to crystallise and refine the crystal structures of all of the homoleptic cages, the disorder induced by the scrambling in the family  $[\text{Cu}_4(\text{MeOL})_{4-x}(\text{CH}_3\text{L})_x]$  ( $x = 1, 2, 3$ ) meant that it was not possible to determine the exact positions of the functional groups on the surface of the cages, which we attribute to the bulky nature of the substituents causing a greater degree of disorder at the periphery of the MOPs. The effect of scrambling was to gradually increase the surface area observed for the cages as  $x$  increased, although the scrambled cages showed lower uptake of  $\text{N}_2$  than the homoleptic cage  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$ . For the family  $[\text{Cu}_4(\text{L})_{4-x}(\text{CH}_3\text{L})_x]$ , it was found that the ligands scrambled across the entire cage, rather than occupying one specific position of the four possible positions in the lantern cage. The increased incorporation of the bulkier methyl group to the structure led to a smooth increase in the unit cell parameter  $c$ , in line with solid solutions. In this case, the scrambled cages showed enhanced surface areas with respect to either of the homoleptic cages  $[\text{Cu}_4(\text{CH}_3\text{L})_4]$  or  $[\text{Cu}_4(\text{L})_4]$ . The scrambled material  $[\text{Cu}_4(\text{L})_2(\text{CH}_3\text{L})_2]$  shows the highest BET surface area determined so far for lantern-type cages. The application of Dubinin–Radushkevich analysis suggests that scrambling leads to materials with a relatively greater contribution to the total pore volume from mesopores



than is found for the pure homoleptic cages. Controlling mesoporosity and the extrinsic porosity of these cages remains a challenge. The contrasting effects on the gas sorption properties of the cages presented here highlight that scrambling approaches are a worthwhile method for MOP synthesis, as they can allow gas uptake to be tuned while retaining a unique pore geometry. This effect could also be studied for scrambling metal sites on the paddlewheels of MOPs, as a means of varying the availability and chemical nature of open metal sites in the activated materials.

## Experimental

### Synthesis

The ligands **LH<sub>2</sub>**, **CH<sub>3</sub>LH<sub>2</sub>**, and **MeOLH<sub>2</sub>** were synthesised following previously reported procedures.<sup>54,56</sup> The solvent content provided is based on the analysis of TGA data and NMR digestions for [Cu<sub>4</sub>(MeOL)<sub>4-x</sub>(CH<sub>3</sub>L)<sub>x</sub>] (*x* = 1, 2, 3) as the solvent could not be refined in the diffraction data; and on the single-crystal X-ray diffraction data for [Cu<sub>4</sub>(L)<sub>4-x</sub>(CH<sub>3</sub>L)<sub>x</sub>] (*x* = 0, 1, 2, 3), where solvent was refined.

[Cu<sub>4</sub>(L)<sub>4</sub>(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 11DMA · 3H<sub>2</sub>O · ([Cu<sub>4</sub>(L)<sub>4</sub>]-DMA). A solution of **LH<sub>2</sub>** (144 mg, 0.30 mmol) in DMA (4 mL) was combined with a solution of Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (58 mg, 0.29 mmol) in DMA (4 mL). The resulting blue solution was placed in an oven at 80 °C overnight, giving 234 mg of blue crystals. 158 mg of these crystals were exchanged with fresh MeOH twice per day for 4 days, affording 114 mg of [Cu<sub>4</sub>(L)<sub>4</sub>] · 5 MeOH as a blue crystalline powder.

The procedure for the synthesis of the scrambled cages was the same in each case, as described in detail here for [Cu<sub>4</sub>(MeOL)<sub>3</sub>(CH<sub>3</sub>L)]-DMA: [Cu<sub>4</sub>(MeOL)<sub>3</sub>(CH<sub>3</sub>L)(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 7DMA · 2H<sub>2</sub>O · ([Cu<sub>4</sub>(MeOL)<sub>3</sub>(CH<sub>3</sub>L)]-DMA). A solution containing **CH<sub>3</sub>LH<sub>2</sub>** (75 mg, 0.15 mmol) and **MeOLH<sub>2</sub>** (240 mg, 0.44 mmol) in DMA (8 mL), was mixed with a solution containing Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (116 mg, 0.58 mmol) in DMA (8 mL). The resulting blue solution was left in the oven at 80 °C overnight giving blue/green crystals that were suitable for single crystal X-ray diffraction. Yield = 478 mg. 304 mg of these crystals were solvent exchanged with fresh MeOH twice per day for 4 days giving 240 mg of [Cu<sub>4</sub>(MeOL)<sub>3</sub>(CH<sub>3</sub>L)] · 5 MeOH as a blue powder. IR spectra, TGA data, PXRD data, and <sup>1</sup>H-NMR spectra of digested samples are provided in the ESI.†

[Cu<sub>4</sub>(MeOL)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 6DMA · 2H<sub>2</sub>O · ([Cu<sub>4</sub>(MeOL)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>]-DMA). **CH<sub>3</sub>LH<sub>2</sub>** (148 mg, 0.29 mmol); **MeOLH<sub>2</sub>** (159 mg, 0.29 mmol); Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (120 mg, 0.60 mmol). Yield = 428 mg. 310 mg of these crystals were solvent exchanged with MeOH giving 249 mg of [Cu<sub>4</sub>(MeOL)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>] · 4 MeOH as a blue powder.

[Cu<sub>4</sub>(MeOL)(CH<sub>3</sub>L)<sub>3</sub>(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 7DMA · 1H<sub>2</sub>O · ([Cu<sub>4</sub>(MeOL)(CH<sub>3</sub>L)<sub>3</sub>]-DMA). **CH<sub>3</sub>LH<sub>2</sub>** (225 mg, 0.44 mmol); **MeOLH<sub>2</sub>** (81 mg, 0.15 mmol); Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (118 mg, 0.59 mmol). Yield = 471 mg. 306 mg of these crystals were solvent exchanged with MeOH giving 240 mg of [Cu<sub>4</sub>(MeOL)(CH<sub>3</sub>L)<sub>3</sub>] · 5 MeOH as blue powder.

[Cu<sub>4</sub>(L)<sub>3</sub>(CH<sub>3</sub>L)(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 8.7DMA · 1.9H<sub>2</sub>O · ([Cu<sub>4</sub>(L)<sub>3</sub>(CH<sub>3</sub>L)]-DMA). **CH<sub>3</sub>LH<sub>2</sub>** (37 mg, 0.07 mmol); **LH<sub>2</sub>** (106 mg, 0.22

mmol); Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (60 mg, 0.30 mmol); DMA (8 mL). Yield = 280 mg. 151 mg of these crystals were solvent exchanged with MeOH giving 117 mg of [Cu<sub>4</sub>(L)<sub>3</sub>(CH<sub>3</sub>L)] · 5 MeOH as a blue powder.

[Cu<sub>4</sub>(L)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 8.6DMA · 1.9H<sub>2</sub>O · ([Cu<sub>4</sub>(L)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>]-DMA). **CH<sub>3</sub>LH<sub>2</sub>** (74 mg, 0.14 mmol); **LH<sub>2</sub>** (70 mg, 0.14 mmol); Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (59 mg, 0.30 mmol); DMA (8 mL). Yield = 222 mg. 143 mg of these crystals were solvent exchanged with MeOH giving 108 mg of [Cu<sub>4</sub>(L)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>] · 5 MeOH as a blue powder.

[Cu<sub>4</sub>(L)(CH<sub>3</sub>L)<sub>3</sub>(DMA)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] · 5.2DMA · 3.5H<sub>2</sub>O · ([Cu<sub>4</sub>(L)(CH<sub>3</sub>L)<sub>3</sub>]-DMA). **CH<sub>3</sub>LH<sub>2</sub>** (111 mg, 0.22 mmol); **LH<sub>2</sub>** (35 mg, 0.07 mmol); Cu(OAc)<sub>2</sub> · H<sub>2</sub>O (60 mg, 0.30 mmol); DMA (8 mL). Yield = 225 mg. 149 mg of these crystals were solvent exchanged with MeOH giving 114 mg of [Cu<sub>4</sub>(L)(CH<sub>3</sub>L)<sub>3</sub>] · 6 MeOH as a blue powder.

### Physical characterisation

Infra-red spectra were collected using a Thermo Scientific spectrometer model NICOLET iS5 using 64 scans and a resolution of 4 cm<sup>-1</sup>. <sup>1</sup>H-NMR spectra were measured with a Bruker AVANCE 400 NMR spectrometer at 25 °C operating at 400.13 MHz for <sup>1</sup>H. For acid digestion of the complexes, *ca.* 15 mg of the complex were suspended in DMSO-d<sub>6</sub> and 40 μL of DCl solution was added. The mixture was left to stand for 3 h at RT, resulting in a yellow solution suitable for NMR measurements. Thermogravimetric analyses (TGA) were performed with a NETZSCH STA 449 F1 Jupiter under N<sub>2</sub> using an isotherm for 10 min at 30 °C before heating up to 500 °C at a rate of 10 °C min<sup>-1</sup>. Powder X-ray diffraction (PXRD) data were collected in a flat plate configuration using a Bruker D8 Discover diffractometer equipped with Cu K<sub>α</sub> source (λ = 1.54056 Å). SEM samples were prepared by placing a small amount of sample on a conductive carbon adhesive label attached to a metal tack, which was sputter coated using a palladium disc connected to a Polar SC500A coater. Coating was carried out for four minutes at a current density of 20 mA under an argon atmosphere. SEM images were taken on a JEOL JSM-IT100 with a Tungsten source and operating at 5 kV and 40 mA current. For gas sorption measurements, all samples were activated *in situ* by heating at 413 K for 10 hours under vacuum before measuring their N<sub>2</sub> uptake at 77 K using a Micrometrics ASAP 2420 (University of Strathclyde) and CO<sub>2</sub> (195 K) using a BELSORP-miniX volumetric adsorption instrument from BEL Japan, Inc (iCeMS, Kyoto University).

### Single crystal X-ray diffraction

All data was collected using a Rigaku model XtaLAB Synergy diffractometer equipped with a Hybrid Pixel Array Detector and Cu K<sub>α</sub> radiation (λ = 1.54184 Å). Structures were solved with the SHELXT solution program using intrinsic phasing, and refined with ShelXL<sup>62</sup> using least squares minimisation (Tables S1–S3†), within the program Olex 2-1.5.<sup>63</sup> Due to high levels of disorder for the family [Cu<sub>4</sub>(MeOL)<sub>4-x</sub>(CH<sub>3</sub>L)<sub>x</sub>] (*x* = 1, 2, 3), only the unit cell parameters are given. Full details of structural refinements are provided in the ESI.† The .cifs have been deposited with the



Cambridge Crystallographic Data Centre (CSD, <https://www.ccdc.cam.ac.uk/>) and have the deposition numbers 2303899 ([Cu<sub>4</sub>(L)<sub>4</sub>]-DMA), 2303900 ([Cu<sub>4</sub>(L)<sub>3</sub>(CH<sub>3</sub>L)]-DMA), 2303901 ([Cu<sub>4</sub>(L)<sub>2</sub>(CH<sub>3</sub>L)<sub>2</sub>]-DMA), and 2303902 ([Cu<sub>4</sub>(L)(CH<sub>3</sub>L)<sub>3</sub>]-DMA).

## Data availability

The data that support the findings of this study are openly available from the University of Strathclyde KnowledgeBase at <https://doi.org/10.15129/abe4cc67-3802-4f83-af0f-d429a799bc3f>.

## Author contributions

BD performed all synthetic procedures and collected IR, NMR, TGA, SEM, PXRD, SXRD, and N<sub>2</sub> sorption data, as well as carrying out initial data interpretation. GAC conceived and supervised the project. ARK finalised the crystal structures and submitted the data to the CSD. ZW and SF collected CO<sub>2</sub> gas sorption isotherms. AJF proposed suitable models and supervised the interpretation of the gas sorption data. All authors discussed the results and commented on the drafts of the manuscript written by BD and GAC.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 T. Tateishi, M. Yoshimura, S. Tokuda, F. Matsuda, D. Fujita and S. Furukawa, Coordination/metal-organic cages inside out, *Coord. Chem. Rev.*, 2022, **467**, 214612.
- 2 S. Lee, H. Jeong, D. Nam, M. S. Lah and W. Choe, The rise of metal-organic polyhedra, *Chem. Soc. Rev.*, 2021, **50**, 528.
- 3 D. Zhang, T. K. Ronson, Y.-Q. Zou and J. R. Nitschke, Metal-organic cages for molecular separations, *Nat. Rev. Chem.*, 2021, **5**, 168.
- 4 A. J. McConnell, Metallosupramolecular cages: from design principles and characterisation techniques to applications, *Chem. Soc. Rev.*, 2022, **51**, 2957.
- 5 T. R. Cook and P. J. Stang, Recent Developments in the Preparation and Chemistry of Metallacycles and Metallacages via Coordination, *Chem. Rev.*, 2015, **115**, 7001.
- 6 S. Saha, I. Regeni and G. H. Clever, Structure relationships between bis-monodentate ligands and coordination driven self-assemblies, *Coord. Chem. Rev.*, 2018, **374**, 1.
- 7 D. J. Tranchemontagne, Z. Ni, M. O'Keeffe and O. M. Yaghi, Reticular Chemistry of Metal-Organic Polyhedra, *Angew. Chem., Int. Ed.*, 2008, **47**, 5136.
- 8 G. A. Craig, P. Larpent, H. Urabe, A. Legrand, M. Bonneau, S. Kusaka and S. Furukawa, Hysteresis in the gas sorption isotherms of metal-organic cages accompanied by subtle changes in molecular packing, *Chem. Commun.*, 2020, **56**, 3689.
- 9 R. A. S. Vasdev, J. A. Findlay, A. L. Garden and J. D. Crowley, Redox active [Pd<sub>(2)</sub>L<sub>(4)</sub>]<sup>(4+)</sup> cages constructed from rotationally flexible 1,1'-disubstituted ferrocene ligands, *Chem. Commun.*, 2019, **55**, 7506.
- 10 L. Chen, T. Yang, H. Cui, T. Cai, L. Zhang and C.-Y. Su, A porous metal-organic cage constructed from dirhodium paddle-wheels: synthesis, structure and catalysis, *J. Mater. Chem.*, 2015, **3**, 20201.
- 11 A. Schmidt, V. Molano, M. Hollering, A. Pothig, A. Casini and F. E. Kuhn, Evaluation of New Palladium Cages as Potential Delivery Systems for the Anticancer Drug Cisplatin, *Chem.-Eur. J.*, 2016, **22**, 2253.
- 12 M. J. Prakash, M. Oh, X. Liu, K. N. Han, G. H. Seong and M. S. Lah, Edge-directed [(M<sub>2</sub>)<sub>2</sub>L<sub>4</sub>] tetragonal metal-organic polyhedra decorated using a square paddle-wheel secondary building unit, *Chem. Commun.*, 2010, **46**, 2049.
- 13 D. A. McMorran and P. J. Steel, The First Coordinatively Saturated, Quadruply Stranded Helicate and Its Encapsulation of a Hexafluorophosphate Anion, *Angew. Chem., Int. Ed.*, 1998, **37**, 3295.
- 14 C. A. Rowland, G. R. Lorzing, A. J. Gosselin, B. A. Trump, G. P. A. Yap, C. M. Brown and E. D. Bloch, Methane Storage in Paddlewheel-Based Porous Coordination Cages, *J. Am. Chem. Soc.*, 2018, **140**, 11153.
- 15 A. López-Olvera, E. Sánchez-González, A. Campos-Reales-Pineda, A. Aguilar-Granda, I. A. Ibarra and B. Rodríguez-Molina, CO<sub>2</sub> capture in a carbazole-based supramolecular polyhedron structure: the significance of Cu(II) open metal sites, *Inorg. Chem. Front.*, 2017, **4**, 56.
- 16 J.-R. Li, A. A. Yakovenko, W. Lu, D. J. Timmons, W. Zhuang, D. Yuan and H.-C. Zhou, Ligand Bridging-Angle-Driven Assembly of Molecular Architectures Based on Quadruply Bonded Mo-Mo Dimers, *J. Am. Chem. Soc.*, 2010, **132**, 17599.
- 17 B. Lee, I.-H. Park and J. Park, Bridging and Fixing Metal-Organic Cages, *ACS Mater. Lett.*, 2022, **4**, 2388.
- 18 J. Park, Z. Perry, Y. P. Chen, J. Bae and H. C. Zhou, Chromium(II) Metal-Organic Polyhedra as Highly Porous Materials, *ACS Appl. Mater. Interfaces*, 2017, **9**, 28064.
- 19 T. H. Chen, L. Wang, J. V. Trueblood, V. H. Grassian and S. M. Cohen, Poly(isophthalic acid)(ethylene oxide) as a Macromolecular Modulator for Metal-Organic Polyhedra, *J. Am. Chem. Soc.*, 2016, **138**, 9646.



- 20 D. Fujita, H. Yokoyama, Y. Ueda, S. Sato and M. Fujita, Geometrically restricted intermediates in the self-assembly of an  $M_{12}L_{24}$  cuboctahedral complex, *Angew. Chem., Int. Ed.*, 2015, **54**, 155.
- 21 M. Tominaga, K. Suzuki, M. Kawano, T. Kusukawa, T. Ozeki, S. Sakamoto, K. Yamaguchi and M. Fujita, Finite, spherical coordination networks that self-organize from 36 small components, *Angew. Chem., Int. Ed.*, 2004, **43**, 5621.
- 22 H. Abourahma, A. W. Coleman, B. Moulton, B. Rather, P. Shahgaldian and M. J. Zaworotko, Hydroxylated nanoballs: synthesis, crystal structure, solubility and crystallization on surfaces, *Chem. Commun.*, 2001, 2380.
- 23 M. Eddaoudi, J. Kim, J. B. Wachter, H. K. Chae, M. O'Keeffe and O. M. Yaghi, Porous Metal–Organic Polyhedra: 25 Å Cuboctahedron Constructed from 12  $Cu_2(CO_2)_4$  Paddle-Wheel Building Blocks, *J. Am. Chem. Soc.*, 2001, **123**, 4368.
- 24 C. F. Espinosa, T. K. Ronson and J. R. Nitschke, Secondary Bracing Ligands Drive Heteroleptic Cuboctahedral  $Pd^{II}_{12}$  Cage Formation, *J. Am. Chem. Soc.*, 2023, **145**, 9965.
- 25 J. A. Findlay, K. M. Patil, M. G. Gardiner, H. I. MacDermott-Opeskin, M. L. O'Mara, P. E. Kruger and D. Preston, Heteroleptic Tripalladium(II) Cages, *Chem.–Asian J.*, 2022, **17**, e202200093.
- 26 C. T. McTernan, J. A. Davies and J. R. Nitschke, Beyond Platonic: How to Build Metal–Organic Polyhedra Capable of Binding Low-Symmetry, Information-Rich Molecular Cargoes, *Chem. Rev.*, 2022, **122**, 10393.
- 27 J. Tessarolo, H. Lee, E. Sakuda, K. Umakoshi and G. H. Clever, Integrative Assembly of Heteroleptic Tetrahedra Controlled by Backbone Steric Bulk, *J. Am. Chem. Soc.*, 2021, **143**, 6339.
- 28 K. E. Ebbert, L. Schneider, A. Platzek, C. Drechsler, B. Chen, R. Rudolf and G. H. Clever, Resolution of minor size differences in a family of heteroleptic coordination cages by trapped ion mobility ESI-MS, *Dalton Trans.*, 2019, **48**, 11070.
- 29 W. M. Bloch, Y. Abe, J. J. Holstein, C. M. Wandtke, B. Dittrich and G. H. Clever, Geometric Complementarity in Assembly and Guest Recognition of a Bent Heteroleptic  $cis-[Pd_2L^A_2L^B_2]$  Coordination Cage, *J. Am. Chem. Soc.*, 2016, **138**, 13750.
- 30 A. Broto-Ribas, M. S. Gutierrez, I. Imaz, A. Carné-Sánchez, F. Gándara, J. Juanhuix and D. Maspoch, Synthesis of the two isomers of heteroleptic  $Rh_{12}L_6L'_6$  metal-organic polyhedra by screening of complementary linkers, *Chem. Commun.*, 2022, **58**, 10480.
- 31 M. Do, D. Rogers, W. Kaminsky and D. J. Xiao, Robust Synthetic Route toward Anisotropic Metal–Organic Cages with Tunable Surface Chemistry, *Inorg. Chem.*, 2021, **60**, 7602.
- 32 N. Hosono, K. Omoto and S. Kitagawa, Anisotropic coordination star polymers realized by self-sorting core modulation, *Chem. Commun.*, 2017, **53**, 8180.
- 33 Y. Fang, X. Lian, Y. Huang, G. Fu, Z. Xiao, Q. Wang, B. Nan, J. P. Pellois and H. C. Zhou, Investigating Subcellular Compartment Targeting Effect of Porous Coordination Cages for Enhancing Cancer Nanotherapy, *Small*, 2018, **14**, 1802709.
- 34 F. R. Dai, U. Sambasivam, A. J. Hammerstrom and Z. Wang, Synthetic supercontainers exhibit distinct solution versus solid state guest-binding behavior, *J. Am. Chem. Soc.*, 2014, **136**, 7480.
- 35 S. Du, C. Hu, J. C. Xiao, H. Tan and W. Liao, A giant coordination cage based on sulfonylcalix[4]arenes, *Chem. Commun.*, 2012, **48**, 9177.
- 36 M. Liu, W. Liao, C. Hu, S. Du and H. Zhang, Calixarene-based nanoscale coordination cages, *Angew. Chem., Int. Ed.*, 2012, **51**, 1585.
- 37 J. Albalad, L. Hernández-López, A. Carné-Sánchez and D. Maspoch, Surface chemistry of metal–organic polyhedra, *Chem. Commun.*, 2022, **58**, 2443.
- 38 E. Sánchez-González, M. Y. Tsang, J. Troyano, G. A. Craig and S. Furukawa, Assembling metal-organic cages as porous materials, *Chem. Soc. Rev.*, 2022, **51**, 4876.
- 39 A. J. Gosselin, C. A. Rowland and E. D. Bloch, Permanently Microporous Metal–Organic Polyhedra, *Chem. Rev.*, 2020, **120**, 8987.
- 40 J. R. Li and H. C. Zhou, Bridging-ligand-substitution strategy for the preparation of metal-organic polyhedra, *Nat. Chem.*, 2010, **2**, 893.
- 41 R. W. Larsen, How Fast Do Metal Organic Polyhedra Form in Solution? Kinetics of  $[Cu_2(5-OH-bdc)_2L_2]_{12}$  Formation in Methanol, *J. Am. Chem. Soc.*, 2008, **130**, 11246.
- 42 S. Walker, W. Chant, K. Tuck and D. Turner, Control of ligand exchange rate between coordination cages by the steric bulk of the ligand and the coordinating solvent, *ChemRxiv*, 2023, preprint, DOI: [10.26434/chemrxiv-2023-95z18](https://doi.org/10.26434/chemrxiv-2023-95z18).
- 43 S. E. Walker, S. A. Boer, T. Malcomson, M. J. Paterson, K. L. Tuck and D. R. Turner, Steric control of sorting regimes in self-assembled cages, *Chem. Commun.*, 2021, **57**, 12456.
- 44 C. von Baeckmann, S. Ruiz-Relaño, I. Imaz, M. Handke, J. Juanhuix, F. Gándara, A. Carné-Sánchez and D. Maspoch, Stepwise assembly of heterometallic, heteroleptic "triblock Janus-type" metal-organic polyhedra, *Chem. Commun.*, 2023, **59**, 3423.
- 45 A. M. Antonio, K. J. Korman, G. P. A. Yap and E. D. Bloch, Porous metal-organic alloys based on soluble coordination cages, *Chem. Sci.*, 2020, **11**, 12540.
- 46 H. Deng, C. J. Doonan, H. Furukawa, R. B. Ferreira, J. Towne, C. B. Knobler, B. Wang and O. M. Yaghi, Multiple Functional Groups of Varying Ratios in Metal–Organic Frameworks, *Science*, 2010, **327**, 846.
- 47 M. P. Rivera, M. Liu, D. He and R. P. Lively, Tuning Material Properties of Porous Organic Cage CC3 with Postsynthetic Dynamic Covalent Chemistry, *Eur. J. Org. Chem.*, 2022, **2022**, e202101507.
- 48 R. L. Greenaway, D. Holden, E. G. B. Eden, A. Stephenson, C. W. Yong, M. J. Bennison, T. Hasell, M. E. Briggs, S. L. James and A. I. Cooper, Understanding gas capacity, guest selectivity, and diffusion in porous liquids, *Chem. Sci.*, 2017, **8**, 2640.



- 49 S. Jiang, J. T. Jones, T. Hasell, C. E. Blythe, D. J. Adams, A. Trewin and A. I. Cooper, Porous organic molecular solids by dynamic covalent scrambling, *Nat. Commun.*, 2011, **2**, 207.
- 50 G. Lal, S. J. Lee, D. M. Spasyuk and G. K. H. Shimizu, Amphiphile-like self assembly of metal organic polyhedra having both polar and non-polar groups, *Chem. Commun.*, 2018, **54**, 1722.
- 51 B. Lerma-Berlanga, J. Castells-Gil, C. R. Ganivet, N. Almora-Barrios, J. González-Platas, O. Fabelo, N. M. Padial and C. Martí-Gastaldo, Permanent Porosity in Hydroxamate Titanium-Organic Polyhedra, *J. Am. Chem. Soc.*, 2021, **143**, 21195.
- 52 A. M. Antonio, K. J. Korman, M. M. Deegan, G. A. Taggart, G. P. A. Yap and E. D. Bloch, Utilization of a Mixed-Ligand Strategy to Tune the Properties of Cuboctahedral Porous Coordination Cages, *Inorg. Chem.*, 2022, **61**, 4609.
- 53 D. Nam, J. Kim, E. Hwang, J. Nam, H. Jeong, T.-H. Kwon and W. Choe, Multivariate porous platform based on metal-organic polyhedra with controllable functionality assembly, *Matter*, 2021, **4**, 2460.
- 54 B. Doñagueta Suso, A. Legrand, C. Weetman, A. R. Kennedy, A. J. Fletcher, S. Furukawa and G. A. Craig, Porous Metal-Organic Cages Based on Rigid Bicyclo[2.2.2]oct-7-ene Type Ligands: Synthesis, Structure, and Gas Uptake Properties, *Chem.-Eur. J.*, 2023, **29**, e202300732.
- 55 Y. Lai, X. He, B. Xue, M. Li, H. Wang, W. Huang, J.-F. Yin, M. Zhang and P. Yin, Modulating Ligand-Exchange Dynamics on Metal-Organic Polyhedra for Reversible Sorting and Hybridization of Mikroarm Star Polymers, *Angew. Chem., Int. Ed.*, 2023, e202311954, DOI: [10.1002/anie.202311954](https://doi.org/10.1002/anie.202311954).
- 56 Z. J. Zhang, W. Shi, Z. Niu, H. H. Li, B. Zhao, P. Cheng, D. Z. Liao and S. P. Yan, A new type of polyhedron-based metal-organic frameworks with interpenetrating cationic and anionic nets demonstrating ion exchange, adsorption and luminescent properties, *Chem. Commun.*, 2011, **47**, 6425.
- 57 T. Hasell, S. Y. Chong, M. Schmidtman, D. J. Adams and A. I. Cooper, Porous organic alloys, *Angew. Chem., Int. Ed.*, 2012, **51**, 7154.
- 58 J. W. M. Osterrieth, J. Rampersad, D. Madden, N. Rampal, L. Skoric, B. Connolly, M. D. Allendorf, V. Stavila, J. L. Snider, R. Ameloot, J. Marreiros, C. Ania, D. Azevedo, E. Vilarrasa-Garcia, B. F. Santos, X. H. Bu, Z. Chang, H. Bunzen, N. R. Champness, S. L. Griffin, B. Chen, R. B. Lin, B. Coasne, S. Cohen, J. C. Moreton, Y. J. Colón, L. Chen, R. Clowes, F. X. Coudert, Y. Cui, B. Hou, D. M. D'Alessandro, P. W. Doheny, M. Dincă, C. Sun, C. Doonan, M. T. Huxley, J. D. Evans, P. Falcaro, R. Ricco, O. Farha, K. B. Idrees, T. Islamoglu, P. Feng, H. Yang, R. S. Forgan, D. Bara, S. Furukawa, E. Sanchez, J. Gascon, S. Telalovic, S. K. Ghosh, S. Mukherjee, M. R. Hill, M. M. Sadiq, P. Horcajada, P. Salcedo-Abraira, K. Kaneko, R. Kukobat, J. Kenvin, S. Keskin, S. Kitagawa, K. I. Otake, R. P. Lively, S. J. A. DeWitt, P. Llewellyn, B. V. Lotsch, S. T. Emmerling, A. M. Pütz, C. Martí-Gastaldo, N. M. Padial, J. García-Martínez, N. Linares, D. MasPOCH, J. A. Suárez Del Pino, P. Moghadam, R. Oktavian, R. E. Morris, P. S. Wheatley, J. Navarro, C. Petit, D. Danaci, M. J. Rosseinsky, A. P. Katsoulidis, M. Schröder, X. Han, S. Yang, C. Serre, G. Mouchaham, D. S. Sholl, R. Thyagarajan, D. Siderius, R. Q. Snurr, R. B. Goncalves, S. Telfer, S. J. Lee, V. P. Ting, J. L. Rowlandson, T. Uemura, T. Iiyuka, M. A. van der Veen, D. Rega, V. Van Speybroeck, S. M. J. Rogge, A. Lemaire, K. S. Walton, L. W. Bingel, S. Wuttke, J. Andreo, O. Yaghi, B. Zhang, C. T. Yavuz, T. S. Nguyen, F. Zamora, C. Montoro, H. Zhou, A. Kirchon and D. Fairen-Jimenez, How Reproducible are Surface Areas Calculated from the BET Equation?, *Adv. Mater.*, 2022, **34**, e2201502.
- 59 M. Thommes, K. Kaneko, A. V. Neimark, J. P. Olivier, F. Rodriguez-Reinoso, J. Rouquerol and K. S. W. Sing, Physisorption of gases, with special reference to the evaluation of surface area and pore size distribution (IUPAC Technical Report), *Pure Appl. Chem.*, 2015, **87**, 1051.
- 60 M. M. Dubinin and L. V. Radushkevich, *Proc. acad. sci. USSR. Phys. chem. sect.*, 1947, **55**, 331.
- 61 M. M. Dubinin, in *Chemistry and Physics of Carbon*, ed. P. L. Walker, Marcel Dekker, New York, 1966.
- 62 G. M. Sheldrick, Crystal structure refinement with SHELXL, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 3.
- 63 O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, OLEX2: a complete structure solution, refinement and analysis program, *J. Appl. Crystallogr.*, 2009, **42**, 339.

