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Insights into molecular accessibility in catalyst and sorbent materials using NMR porosity measurements†

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NMR measurements of porosity are demonstrated for a range of catalyst and sorbent materials. The NMR method allows porosity measurements to be carried out directly with the molecules of interest for a given reaction or separation process, thereby providing valuable information regarding accessibility of that molecular species to the pore network during process operation.

The porosity, ϵ , of a solid material is an important characterization parameter that can be used to understand both the mechanical properties of a material¹ and the transport properties of fluids confined within it.^{2,3} As such, porosity is a key parameter to control when optimising the properties of a wide range of heterogeneous systems. Many techniques already exist to measure porosity, but the most common methods still possess significant drawbacks. For example, the high pressures used for mercury porosimetry are well-known to damage pore structures,⁴ it cannot detect pores that are smaller than a few nm, the pore shielding effect can lead to inaccurate pore size distributions,⁵ and it is a destructive technique due to the inability to fully remove mercury from the sample after analysis.⁶ Helium pycnometry must be combined with an additional bulk or pore volume measurement to calculate porosity and can become erroneous for materials which adsorb helium, such as molecular sieves measured at high pressures.⁷ Furthermore, both methods rely on the use of indirect probe molecules to measure the accessible porosity.⁶ The use of indirect probe molecules may give misleading results if the probe is able to access different parts of the pore network compared to the actual fluids of interest. For example, in cases where the pore size is comparable to the size of the probe molecule even subtle changes in the structure can lead to large differences in the

adsorption capacity, as was shown for the four-fold greater adsorption of toluene in EU type zeolites compared to the adsorption of 1,3,5-trimethyl benzene.⁸ The size selectivity of microporous materials such as zeolites makes it challenging to interpret porosity measurements that rely on indirect probe molecules such as He, N₂, Ar, or Hg. NMR offers an opportunity to overcome this limitation as it can be implemented with any probe molecule that contains a nuclear spin with non-zero nuclear spin quantum number. The vast majority of existing NMR porosity methods use water as the probe molecule. CPMG measurements have been used extensively to measure total porosity and pore size distributions in macroporous media, including rock cores^{9–11} and cements.¹² Intensity differences in MRI images between intra- and extra-particle liquid have been used to provide spatially resolved porosities,¹³ which agreed with gravimetric measurements to within 2% for rock core systems, and NMR cryoporometry measurements¹⁴ have been applied to measure the nanoporous pore size distribution and total porosity of titania films¹⁵ and silica foams.¹⁶ Typically, NMR porosity measurements require an independent measurement or estimate of the bulk volume of the porous material to yield a porosity value. However, in recent years, techniques based on three separate NMR measurements of intra-pellet fluid, intra- and extra-pellet fluid, and bulk fluid have been applied to measure the porosity of irregular shaped porous media without the need for independent measurements of porosity.^{17,18} Dick *et al.* have coupled such an approach with a CPMG porosity measurement for studying the porosity of rock core cuttings¹⁷ and Linck *et al.* used ¹H spectroscopy to measure the porosity of silica spheres.¹⁸ For the latter, good agreement between NMR porosity measurements and gas sorption porosity measurements was observed for the large pore silicas, but much poorer agreement was observed for water in the smallest pore size silicas, with NMR underestimating the porosity by up to 9%. Further

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consideration of the factors contributing to the quantitative nature of the NMR measurement are required to ensure a robust implementation of these NMR methods for catalyst and sorbent materials, and to allow deeper physicochemical understanding of liquid accessibility.

Herein we present a simple and robust NMR porosity measurement that allows the apparent porosity experienced by any molecule containing NMR-active nuclei within a porous material to be directly measured. It is therefore possible to measure the porosity experienced by the reactants, products and sorbates of interest, with the scope to apply this method at the temperatures and pressures relevant to the chemical process under investigation. In the present work, porosity measurements are made using ^1H nucleus observation. The method is applied to a range of porous oxides and zeolites initially using water as the probe molecule. When water is used as the probe molecule it is not necessary to dry the sorbent prior to analysis, making the NMR porosity method a rapid and simple characterisation tool. The method is then repeated with a range of different probe molecules to demonstrate the differences in apparent porosity that can be observed for different molecules, these insights could not be obtained unambiguously from conventional porosity measurements.

The principles of the measurement method are now given. The porosity of a material can be defined as:

$$\varepsilon = \frac{V_p}{V_p + V_s}, \quad (1)$$

where V_p is the volume of the pore space and V_s is the volume of the solid matrix. The NMR porosity method introduced in this work calculates the porosity by measuring the signal intensities of three NMR spectra corresponding to: (1) saturated pellets (intra-pellet fluid, giving rise to NMR signal S_1), (2) immersed pellets (intra- and extra-pellet fluid giving rise to signal S_2), and (3) bulk fluid without pellets (signal S_3). The absolute signal intensity, S_i , of each NMR spectrum is given by:¹⁹

$$S_i = K_s N_i, \quad (2)$$

where K_s is a spectrometer constant that is influenced by the choice of hardware and parameter selection, N_i is the number of ^1H spins contained within the sample, and i is a label denoting which of the three NMR spectra is being referred to. A relaxation correction (see ESI†) is applied to each signal to avoid introducing systematic errors due to unequal rates of signal decay between samples and, in particular, due to the rapid relaxation observed for liquids in porous media. Under properly optimized conditions K_s will be constant for all spectra acquired on the same spectrometer – however for NMR challenging samples (*e.g.* high metal loadings) this may require further validation. The volume occupied by the ^1H species in the measurement region (*i.e.* the radiofrequency, RF, coil) is given by:

$$V_i = \frac{S_i V_m}{N_A K_s n_p}, \quad (3)$$

where V_m is the molar volume of the liquid, N_A is Avogadro's constant, and n_p is the number of ^1H species per molecule of the probe liquid. S_1 is proportional to the volume of the porous material that is accessible to the probe molecule and $S_3 - S_2$ is proportional to the inaccessible volume of the material. The value of K_s will likely change at the extremities of the coil. By having samples 2 and 3 extend beyond the top of the RF coil and by limiting sample 1 to the homogeneous region of the RF coil, this source of error is minimised (see ESI† for further details).

It follows that, when the size of the pores far exceeds the size of the probe molecules, the accessible and inaccessible volumes are well approximated by V_p and V_s respectively. Under these conditions and the assumption of a constant molar volume upon adsorption, eqn (1) and (3) can be combined to give:

$$\varepsilon = \frac{S_1}{S_1 + S_3 - S_2}. \quad (4)$$

However, when the size of the probe molecules exceeds the size of the pore opening, part of the pore space becomes inaccessible, and eqn (4) then measures the apparent porosity experienced by the probe molecule, which may vary significantly from the open structural porosity. This approach to measure porosity has significant advantages over the NMR CPMG-based T_2 relaxation time methods used, for example, in petrophysical analysis.^{20,21} In particular, the proposed method allows the characterisation of porous materials in which the NMR relaxation time constants are shorter than the NMR CPMG echo time required for a T_2 measurement (*e.g.* microporous materials); under such conditions the T_2 measurement cannot be employed due to rapid signal loss. Further, the proposed method does not require an independent measurement of the bulk volume.

NMR experiments were carried out using a Bruker AV400 NMR spectrometer operating at a ^1H frequency of 400.23 MHz, equipped with a Diff30 probe and 5 mm $^1\text{H}/^{13}\text{C}$ coil. Fig. 1 shows an excellent agreement between the NMR porosity measurement and conventional measurements when applied to a range of titania, alumina, and type A zeolite pellets. Here water was used as the probe and Fig. 1 shows that the NMR porosities were consistent to within $\pm 2\%$ of the conventional measurements. By choosing water as the probe molecule it is possible to remove sample drying protocols and the measurements here were taken in less than 15 min per sample. The method is, in theory, extendable to powder samples noting that due care must be taken to remove extra-particle liquid.

The conventional porosity measurements were calculated as:²²

$$\varepsilon = 1 - \frac{\rho_s}{\rho_b}, \quad (5)$$

where ρ_b is the bulk density from mercury porosimetry and ρ_s is the skeletal density of the catalyst pellets. For the aluminas



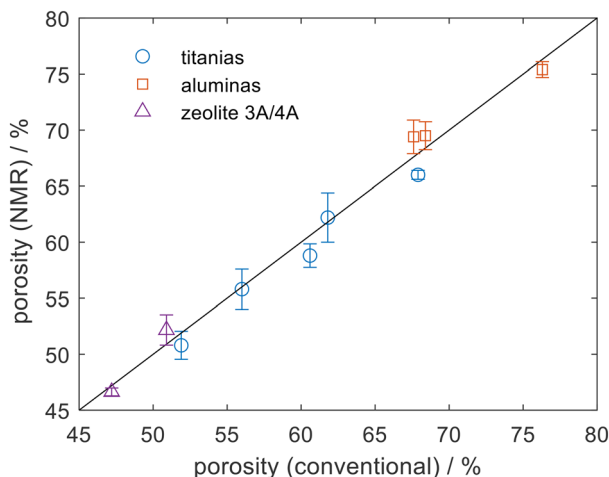


Fig. 1 A comparison between the NMR porosity measurement and mercury porosimetry measurements applied to a range of titanias and aluminas. Skeletal densities of zeolites 3A and 4A were measured with helium pycnometry to ensure that the zeolite microporosity was captured. The solid line is a line of $y = x$. Errors bars represent the standard error of 3 measurements.

and titanias, argon physisorption results showed negligible contributions from pores <5 nm and mercury porosimetry was used to measure the skeletal density. For zeolites 3A and 4A, argon physisorption data were consistent with significant microporosity as well as meso/macroporous regions detected by mercury porosimetry, as shown in Fig. 2 for zeolite 4A. In addition to the complex pore size distribution observed by mercury porosimetry and argon physisorption there is an expected ultramicroporosity corresponding to the zeolite crystals themselves that is not well detected by either technique.^{6,23}

To ensure that the total porosity was being detected both mercury porosimetry and helium pycnometry were carried out for the zeolite 3A and 4A samples. The complex pore size

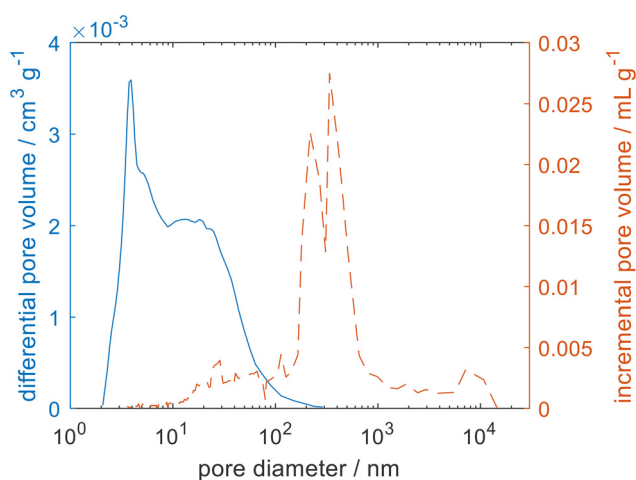


Fig. 2 The pore size distribution measured for zeolite 4A using argon physisorption (blue solid line) and mercury porosimetry (orange dashed line).

distributions can be attributed to a hierarchical pore structure. For zeolite 4A, the zeolite crystals have a nominal pore opening of 4 Å. Surrounding the zeolite crystals is a porous network spanning several orders of magnitude of pore size, associated with the intercrystal space and the binder used to pelletise the zeolite beads, typical of such materials.²⁴ Helium pycnometry gave a much larger porosity for the zeolites ($\epsilon_{HP} = 50.9\%$), when compared to mercury porosimetry ($\epsilon_{MP} = 32.9\%$). As mercury porosimetry is unable to access the microporosity, the large difference between the two measurements is consistent with a microporosity of 18%. Such a significant difference between ϵ_{HP} and ϵ_{MP} has previously been observed for similar materials that show a broad distribution of pore sizes and significant microporosity such as mudrocks²⁵ and cements.²⁶ It follows that the observed porosity will be a function of the size of the chosen probe molecule and that it is not always trivial to relate ϵ_{HP} and ϵ_{MP} to the porosity experienced by the actual sorbate-sorbent pairs of interest.

The major advantage of the NMR porosity method is that it can perform chemically-specific measurements and therefore can be used to directly measure the porosity of the sorbate-sorbent system of interest. It can also be used to measure porosity at relevant conditions of temperature and pressure. Fig. 3 shows porosity measurements for 3 materials: an alumina (Al-I) and zeolites 3A and 4A using 5 different probe molecules. Each material was dried (see Fig. S4 in ESI†) and the porosity measurements were repeated with each different probe molecule. Fig. 3 shows the porosity for each material as a function of the critical diameter of the probe molecule. For these experiments the critical diameter was defined as the second smallest molecular dimension of each molecule from ZINDO calculations,^{27,28} which would be the limiting dimension for accessing a circular pore opening

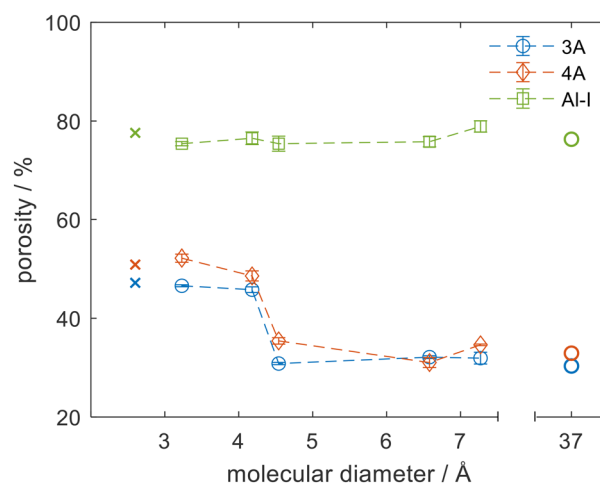


Fig. 3 The porosity experienced by different probe molecules in alumina and zeolites 3A and 4A. The probe molecules are (left to right on the dashed line) water, methanol, *n*-decane, cyclohexane, and *o*-xylene. Helium pycnometry (x) and mercury porosimetry (o) results are shown for comparison.



of the zeolite crystals. The porosities measured by NMR were compared to helium pycnometry and mercury porosimetry methods, which are denoted as having critical diameters of 2.6 and 37 Å respectively. For helium pycnometry this value relates to the kinetic diameter of helium gas²⁹ and for mercury porosimetry it corresponds to the pore size accessed at the highest pressure of the mercury porosimetry experiment used in these analyses. For the mesoporous alumina, Al-I, the pore size distribution was monomodal with an average pore size of 12 nm. The pores were far larger than the critical diameter of all probe molecules and no variation in porosity was noted as the probe molecule was varied. Excellent agreement was observed between the NMR method and both the helium pycnometry and mercury porosimetry methods for Al-I, which indicated that all molecules were able to access the full porosity. In contrast, the zeolite materials showed a significant decrease in porosity as the size of the probe molecule increased, with a step change from a high porosity region to a low porosity region at 4.2 Å. The change in effective porosity is consistent with a change in accessibility of the probe molecule. When the size of the probe molecule was smaller than the aperture size of the zeolite pores the probe molecule could access both the intracrystal zeolite porosity and the binder porosity, leading to a high value of porosity (48% and 52% for 3A and 4A respectively). When the size of the probe molecule exceeded that of the zeolite aperture then the probe molecule was excluded from the microporosity, and only the porosity of the intercrystalline space was measured (31% and 35% for 3A and 4A respectively). These arguments agree with the helium pycnometry and mercury porosimetry data.

The sharp transition in porosity was expected for well-controlled zeolitic materials such as the 3A and 4A molecular sieves. The position of the transition, however, was not simple to predict *a priori*. Methanol (critical diameter of 4.2 Å) was shown to fully access the microporosity of zeolites 3A and 4A despite being slightly larger than the nominal pore size of the latter and much larger than the former. This result is consistent with previous drying (*i.e.* water removal) studies of primary alcohols, which suggested that competitive adsorption within the 3A crystals led to poorer drying of methanol by 3A zeolites compared to the drying of larger alcohols and esters, which could not access the zeolite microporosity.³⁰ The data shown in Fig. 3 support the argument that as both water and methanol can enter the zeolite cages, competitive adsorption between water and methanol plays an important role in the drying process rather than a pure size exclusion of the methanol as may have been expected based on the zeolite pore sizes and molecular diameters. Such physicochemical insights are only possible when the species of interest are used as probe molecules themselves at relevant process conditions, as complex effects such as chemical interaction and structural flexing are not accurately detected with indirect probe molecules. Small changes in the system can result in significant changes in behaviour. For example, previous work

has shown that when methanol adsorption on zeolite 3A is carried out in the gas phase rather than the liquid phase, poor methanol adsorption is observed.^{31,32} Conflicting literature results coupled with discrepancies in definitions of molecular diameters, differences in zeolite compositions,³³ breathing phenomena³⁴ and interactions between adsorbates and cations leading to trapdoor effects³⁵ all complicate the process of predicting molecular accessibility computationally or from indirect measurements, and highlight the need for molecule-specific measurements of molecular accessibility.

In summary, a simple NMR methodology for measuring porosity has been outlined, and it has been demonstrated that by ensuring RF homogeneity of the magnetic field in which the NMR measurement is made and by quantitatively accounting for T_2^* relaxation, agreement between magnetic resonance measurements of porosity and conventional measurements was reduced to <2% whilst also having the advantage of allowing rapid molecule-specific porosity values at any conditions of temperature. The chemical-specific nature of the measurement has been demonstrated in a case study in which the porosity of zeolites 3A and 4A was measured as a function of probe molecule. A significant decrease in porosity was observed between methanol and *n*-decane, corresponding to the latter being unable to access the microporosity of the zeolite crystals. Methanol was shown to access the micropores of zeolite 3A despite being significantly larger than the nominal pore opening. This result provides further evidence that methanol drying with zeolites 3A is not driven purely by size exclusion as is often assumed. For microporous systems a direct measurement of the porosity each molecule experiences is key to understanding the physicochemical process that occur within them, rather than relying on porosity measurements using indirect probes and calculations of molecular diameters. These measurements have been made on a 400 MHz spectrometer and using ¹H nucleus observation. However, the measurements can be made using low field magnet hardware, time domain NMR methods, and nuclei other than ¹H, as long as there is sufficient signal-to-noise. Therefore, the approach offers the potential for widespread use in research and analytical laboratories.

Author contributions

JWW: conceptualization, investigation, methodology, writing – original draft. CMG: conceptualization, investigation, resources, funding acquisition. MDM: methodology, supervision, writing – review and editing. AJS: methodology, supervision. LFG: conceptualization, funding acquisition, supervision, writing – review and editing.

Conflicts of interest

There are no conflicts to declare.



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Notes and references

- M. Zakeri, A. Samimi, M. Shafiee Afarani and A. Salehirad, *Part. Sci. Technol.*, 2018, **36**, 96–103.
- A. Koponen, M. Kataja and J. Timonen, *Phys. Rev. E*, 1997, **56**, 3319–3325.
- P. Sudarsanam, E. Peeters, E. V. Makshina, V. I. Parvulescu and B. F. Sels, *Chem. Soc. Rev.*, 2019, **48**, 2366–2421.
- X. Wang, Y. Peng, J. Wang and Q. Zeng, *Materials*, 2019, **12**, 2220.
- S. P. Rigby and L. F. Gladden, *Chem. Eng. Sci.*, 2000, **55**, 5599–5612.
- L. Espinal, in *Characterization of Materials*, American Cancer Society, 2012, pp. 1–10.
- A. Arami-Niya, T. E. Rufford, G. Birkett and Z. Zhu, *Microporous Mesoporous Mater.*, 2017, **244**, 218–225.
- Z. Guo, W. Hao, J. Ma and R. Li, *J. Chem. Res.*, 2020, **45**, 187–193.
- P. Zhao, B. He, B. Zhang and J. Liu, *Pet. Sci.*, 2022, **19**, 509–517.
- M. Appel, *Petrophysics – The SPWLA Journal of Formation Evaluation and Reservoir Description*, 2004, vol. 45.
- J. R. Hook, *Petrophysics – The SPWLA Journal of Formation Evaluation and Reservoir Description* 2003, vol. 44.
- A. C. A. Muller, K. L. Scrivener, A. M. Gajewicz and P. J. McDonald, *J. Phys. Chem. C*, 2013, **117**, 403–412.
- F. Marica, Q. Chen, A. Hamilton, C. Hall, T. Al and B. J. Balcom, *J. Magn. Reson.*, 2006, **178**, 136–141.
- J. Mitchell, J. B. W. Webber and J. H. Strange, *Phys. Rep.*, 2008, **461**, 1–36.
- D. Vargas-Florencia, T. Edvinsson, A. Hagfeldt and I. Furó, *J. Phys. Chem. C*, 2007, **111**, 7605–7611.
- D. Vargas-Florencia, I. Furó and R. W. Corkery, *Langmuir*, 2008, **24**, 4827–4832.
- M. J. Dick, D. Veselinovic, T. Kenney and D. Green, *Validation of Porosity of NMR Cuttings. Paper presented at the SPWLA 63rd Annual Logging Symposium, Stavanger, Norway, June 2022.*
- L. G. Linck, S. A. Maldonado Ochoa, M. Ceolín, H. Corti, G. A. Monti, F. V. Chávez and R. H. Acosta, *Microporous Mesoporous Mater.*, 2020, **305**, 110351.
- S. K. Bharti and R. Roy, *TrAC, Trends Anal. Chem.*, 2012, **35**, 5–26.
- Y. Yao, D. Liu, Y. Che, D. Tang, S. Tang and W. Huang, *Fuel*, 2010, **89**, 1371–1380.
- J. Lai, G. Wang, Z. Fan, J. Chen, S. Wang, Z. Zhou and X. Fan, *Energy Fuels*, 2016, **30**, 10200–10214.
- L. E. Flint and A. L. Flint, *Methods of Soil Analysis*, 2002, pp. 241–254.
- M. Thommes, K. Kaneko, A. V. Neimark, J. P. Olivier, F. Rodriguez-Reinoso, J. Rouquerol and K. S. W. Sing, *Pure Appl. Chem.*, 2015, **87**, 1051–1069.
- L. Gueudré, M. Milina, S. Mitchell and J. Pérez-Ramírez, *Adv. Funct. Mater.*, 2014, **24**, 209–219.
- B. Andreas, S. Kevin, K. Niko, C. Ab, P. Vitaliy, F. Artem, L. Leon, A.-H. Alexandra and B. Pieter, *Spec. Publ. - Geol. Soc. London*, 2017, **454**, 15–38.
- T. Tracz and T. Zdeb, *Materials*, 2019, **12**, 192.
- C. E. Webster, R. S. Drago, M. C. Zerner and V. Gaines, *J. Am. Chem. Soc.*, 1998, **120**, 5509–5516.
- B. Chen, Y. Ji, M. Xue, F. R. Fronczek, E. J. Hurtado, J. U. Mondal, C. Liang and S. Dai, *Inorg. Chem.*, 2008, **47**, 5543–5545.
- D. W. Breck and D. W. Breck, *Zeolite molecular sieves: structure, chemistry, and use*, John Wiley & Sons, 1973.
- C. Pahl, C. Pasel, M. Luckas and D. Bathen, *J. Chem. Eng. Data*, 2012, **57**, 2465–2471.
- G. Pascual-Muñoz, R. Calero-Berrocal, M. Larriba, V. I. Águeda and J. A. Delgado, *Microporous Mesoporous Mater.*, 2023, **360**, 112669.
- P. Azhagapillai, M. Khaleel, F. Zoghieb, G. Luckachan, L. Jacob and D. Reinalda, *ACS Omega*, 2022, **7**, 6463–6471.
- R. Lin, A. Ladshaw, Y. Nan, J. Liu, S. Yiacoumi, C. Tsouris, D. W. DePaoli and L. L. Tavlarides, *Ind. Eng. Chem. Res.*, 2015, **54**, 10442–10448.
- H. Shi, A. N. Miguez and S. M. Auerbach, *Green Chem.*, 2014, **16**, 875–884.
- J. Shang, G. Li, R. Singh, Q. Gu, K. M. Nairn, T. J. Bastow, N. Medhekar, C. M. Doherty, A. J. Hill, J. Z. Liu and P. A. Webley, *J. Am. Chem. Soc.*, 2012, **134**, 19246–19253.

