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1. Introduction

Polymer materials are ubiquitous in daily life with their applications in packaging,¹ coating,² construction,³ medical,⁴ electronics,⁵ energy storage,⁶ and catalysis.⁷ Over 400 million tons of polymer materials were produced worldwide in 2021.^{8,9} Polymer materials have drawn great concerns because of their environmental and sustainable issues. Current concerns mainly focus on the degradation and the recycling of post-consumer polymer waste.¹⁰⁻¹² Very few studies focus on the sustainability of polymer production. Many commonly used vinyl polymers are synthesized through free radical polymerizations. The initiation of free radical polymerization usually requires elevated temperatures (60 °C or above) for thermal initiators, such as 2,2'-azobisisobutyronitrile (AIBN), to decompose.13 Considering the volume of polymers produced each year, the energy consumed by hightemperature polymerization techniques is enormous.14,15 With the Paris Agreement, improving energy efficiency in industries is critical for reducing the emissions of greenhouse gases and improving their carbon footprint.¹⁶

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Synthesis of acrylonitrile–butadiene–styrene copolymers through interface-initiated room-temperature polymerization†

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Acrylonitrile–butadiene–styrene (ABS) copolymers were synthesized in emulsions and silica-containing emulsion gels at 20, 40, and 60 °C. The room-temperature polymerization was achieved by decomposing 2,2'-azobisisobutyronitrile (AIBN) at oil–water interfaces. The decomposition rate constants of the AIBN decomposed in bulk phases and at interfaces were measured for the first time. At room temperatures, the decomposition of AIBN primarily occurred at oil–water interfaces. At 60 °C, the decomposition of AIBN occurred both in bulk phases and at interfaces. In a dark environment without inert gas protection, the polymerizations in emulsions and emulsion gels were conducted at room temperatures successfully. The presence of fumed silica particles enhanced the interfacial initiation and the subsequent polymerization. In the presence of fumed silica, the polymerizations at 40 °C can be as fast as that conducted without silica at 60 °C. The molar mass of the ABS copolymers increased with decreased polymerization temperatures. The ABS copolymers with an ultra-high molar mass and narrow molar mass distribution were synthesized. The ABS copolymers with ultra-high molecular masses exhibit improved ductility and thermal properties without compromising Young's modulus and surface hardness. Interfacial initiation is an eco-friendly technique to produce high-performance polymer materials.

Conducting free radical polymerization at room temperature can be achieved through several approaches. The most straightforward approach is using a thermal initiator that can be decomposed at room temperature. Perfluorodiacyl peroxides, trialkyl boranes, and some azo compounds can produce radicals efficiently at room temperature, but these low-temperature initiators require strict regulation for storage and use.^{17,18} Redox initiators can be an effective initiation approach at room temperature. Still, their applications are limited due to the presence of salts and transitional metal compounds.^{15,16,19} The initiation using radiation, such as UV light and γ -ray, is limited by their penetration depth and production scale.^{20,21} Recently, oil-water interfaces have been found to accelerate some chemical reactions.²² Surfactantstabilized oil-water interface processes chemical and physical properties that are different from bulk, such as interfacial tension,²³ polarity or reactant gradient,²⁴⁻²⁶ steric effect,^{27,28} complex formation,29 and electrostatic activities.30 Some of these interfacial properties may enhance the reaction kinetics at oil-water interfaces.^{22,27,28,31,32} In a dark environment at room temperature, many commonly used thermal initiators, such as AIBN and potassium persulfate, can decompose at oilwater interfaces or through interfacial initiation at room temperature.13,23,32 In light of interfacial initiation, the oilwater interfaces in emulsions can be catalytic sites where thermal initiators can decompose at low temperatures. As a



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result, a low-temperature radical polymerization is achievable through interfacial initiation in emulsion systems to improve energy efficiency.

Acrylonitrile-butadiene-styrene (ABS) copolymer is an important engineering thermoplastic and is widely used in automotive, electronics, and membranes.^{33,34} The acrylonitrile/styrene segments make the ABS copolymer mechanically strong and resistant to chemicals while the butadiene segments adjust their toughness.³³ The global production of ABS materials was 14 million tons in 2022 and was projected to increase by 37% by 2027.³⁵ Emulsion polymerization is one of the favorite methods to obtain ABS copolymer materials.³³ In light of interfacial initiation, the energy consumption of ABS copolymers and their composite materials can be reduced by decreasing the polymerization temperatures in emulsion systems.

Herein, we report the synthesis of ABS copolymer composite monoliths using AIBN as an initiator in emulsions or silicacontaining emulsion gels through interfacial initiation. Polymer monoliths are important engineering materials used in separation, extraction, medical, and catalysis.^{36,37} The polymerizations were conducted at the normal decomposition temperature for AIBN (60 °C) or the lower-than-normal temperatures (20 and 40 °C). The kinetics for initiation and polymerization, the structural characterization, and the mechanical properties of the ABS monoliths synthesized at different temperatures were investigated.

2. Materials and methods

Styrene, acrylonitrile, and 1,3-butadiene were purchased from Aldrich, Adamas, and TCI, respectively. The stabilizers in the monomers were removed by passing through a basic aluminapacked column. Cetyltrimethylammonium bromide (CTAB) was obtained from Aldrich. 2,2'-Azobisisobutyronitrile (AIBN) (Adamas) was purified by recrystallization in methanol before use. Fumed silica Cab-O-Sil M5 with a diameter of 20 nm was provided by Cabot. Toluene, dodecane, 1-butanol, and deuterated chloroform were purchased from Adamas.

Silica-containing emulsion gels were prepared using the following procedure: 24 ml monomers (acrylonitrile/1,3-butadiene/styrene 30/10/60 by volume), 1.2 g fumed silica, and 0.5 g AIBN were mixed first in a glass vial using a vortex mixer. 6 ml CTAB aqueous solution (0.5 M) was then added, and the mixtures were shaken for 1 min to form stable emulsion gels. The appearance of the silica-containing emulsion gels can be found in the literature.^{38,39} Emulsion samples were prepared using the same procedure without adding fumed silica. All the samples were capped and sealed with Teflon tape.

The rheological properties of the emulsion gels were tested using an Anton Paar MCR 302e rheometer with parallel plates. For each sample, the monomers were replaced by a high-boiling point organic solvent dodecane, and 0.5 mL materials were used for the testing. A steady shear was applied before each rheological test.

The polymerizations were conducted in a dark environment to eliminate the possibility of UV exposure. The room-temperature polymerizations were carried out at laboratory temperature, 20 °C, and the high-temperature polymerizations (40 and 60 °C) were conducted in an oil bath. At different times, aliquots of the polymerizing samples were taken out and dried under vacuum to remove water and unreacted monomers. The conversions of monomers were calculated by subtracting the masses of fumed silica and CTAB from the masses of the dried samples.

Pure polymers were extracted from the polymerized monoliths by dissolving them in dimethylformamide at 60 °C. The undissolved portion (fumed silica and CTAB) was discarded. The extracted polymers were dried and redissolved in tetrahydrofuran to measure molar mass and dispersity. Gel permeation chromatography (GPC) measurement was carried out using Waters E2695 with a Heleos light-scattering detector and an Optilab rEX refractive index detector.

For the decomposition of AIBN, the compositions of the emulsion and emulsion gel samples are the same as the polymerizing sample, except the monomers were replaced by non-polymerizable toluene. At different times, portions of the samples were taken out and converted to microemulsions by adjusting the composition to their microemulsion range, *i.e.*, toluene/CTAB/water/1-butanol 55/15/9/21 by mass.³² The absorption spectra for the microemulsions were characterized using an Agilent Cary 60 UV/visible spectrophotometer and the absorption at 344–346 nm was recorded for azo absorption.³² The UV spectra for the bulk samples (toluene/AIBN) were measured directly without converting to microemulsions.

The morphologies of the as-synthesized ABS materials were characterized by Hitachi Regulus 8100 scanning electron microscope (SEM). The Fourier transform infrared spectroscopy (FTIR) was recorded using a Bruker Invenio S spectrometer. The nuclear magnetic resonance (NMR) spectrum was tested using a Bruker Avance Core 400 MHz spectrometer with deuterated chloroform. The differential scanning calorimetry (DSC) thermograms were measured using a TA DSC 25 with a ramping rate of 10 °C min⁻¹, and the thermograms from the second heating scan were recorded. The mechanical properties of the as-synthesized monoliths were tested using an Anton Paar UNHT nanoindentation tester with a maximum load of 10.00 mN, loading/unloading rate of 60.0 mM min⁻¹, and a pause of 5.0 s. For each sample, at least 5 indents were performed, and the average values were reported.

3. Results and discussion

3.1 Initiation at interfaces

AIBN as a commonly used thermal initiator decomposes at 60 °C or above to generate two radicals and release one nitrogen. The decomposition of AIBN in toluene, emulsion, and emulsion gel samples was measured from the azo-absorption in UV spectra, and the decomposition rate constant (k_d) was calculated based on the following equation:

$$A_t = A_0 e^{(-k_{\rm d}t)}$$

where A_t and A_0 are the absorbance for the azo compounds at time *t* and *t* = 0, respectively. As seen in Fig. 1, the AIBN barely decomposed in toluene at 20 °C and the decomposition occurred very slowly at 40 °C. Both observations are consistent with the results reported by others.^{32,40,41} At 60 °C, AIBN decomposed freely in toluene as the thermal energy overcame the activation energy of AIBN decomposition. The measured rate constant, $7.21 \times 10^{-5} \text{ s}^{-1}$, is in line with the values reported by others.^{32,42}

Interestingly, the decomposition of AIBN at 20 °C was observed in emulsion systems. AIBN is insoluble in water and barely decomposes in toluene at this low temperature. The observed decomposition in the emulsions most likely originated from the surfactant-stabilized oil-water interfaces. Due to the heterogeneous environment and the complicated initiation process in emulsions,^{43,44} the decomposition of AIBN contributed from interfaces is difficult to measure directly. By assuming that the decomposition of AIBN in bulk phases and at interfaces are two independent processes, the decomposition rate constants of AIBN at interfaces were calculated using the following equation:

$$k_{\rm d,t} = k_{\rm d,b} + k_{\rm d,i}$$

where $k_{d,t}$, $k_{d,b}$, and $k_{d,i}$ are the decomposition rate constant in total, in toluene (bulk phase), and at interfaces, respectively. The measured and calculated decomposition rate constants are listed in Table 1. The decomposition of AIBN contributed from interfaces in emulsion systems becomes clear. At 20 °C, AIBN did not decompose in bulk phases, so the observed decomposition of AIBN in emulsion systems are contributed from the interfaces solely, which is 3.1×10^{-7} s⁻¹ for emulsions. The decomposition of AIBN at interfaces became more pronounced at 40 °C. The values of the $k_{d,i}$ were found to be 3.7 times greater than that of $k_{d,b}$, suggesting the decomposition of AIBN primarily occurs at oil-water interfaces while very few were observed for bulk phases at low temperatures. At 60 °C, the $k_{d,b}$ and the $k_{d,i}$ were further increased and the measured values were close to each other.

The reaction temperatures for some aqueous and organic phase reactions can be lowered at oil-water interfaces, consequently accelerating the reaction rates.^{22,26} In these cases, the catalytic effect was achieved through the accumulation of reactants at or near the oil-water interfaces or within micelles, resulting in a higher localized reactant concentration.^{25,26} reactions involving multiple The components therefore can be catalyzed through this mechanism or so-called micellar catalysis. AIBN decomposes through a homolytic cleavage of its own C-N bonds, which should not be sensitive to the concentration of AIBN.¹³ In fact, the decomposition temperature for AIBN was slightly lowered if they were diluted in an organic solution.40 Although AIBN initiators prefer to migrate to the oil-water interfaces in emulsions,²³ the concentration of AIBN at the interfaces is unlikely to be the reason for the lowered decomposition temperature observed in this study.

In addition to the concentration effects, the interfacial energy of the oil-water interfaces and the complex formation may be the other possible mechanisms. The former is based on the fact that the interfacial energy of the micelles are 200 times greater than the thermal energy needed for AIBN to decompose.²³ With the assistance of the additional interfacial energy, AIBN should readily decompose on the micelle surfaces at room temperature. The latter proposed the formation of a CTAB-AIBN complex roughly in a 3:1 molar ratio, which lowered the activation energy for AIBN decomposition.²⁹ Both theories seem to explain the decomposition of AIBN in emulsions at 20 °C.

By increasing the temperature to 40 °C, the observed decomposition of AIBN in both bulk phases and at interfaces was enhanced. The higher the temperature, the more thermal energy was provided to the molecules in the emulsion systems. The slow decomposition of AIBN in toluene at 40 °C is agreed with the onset decomposition temperature of AIBN in the same conditions measured using calorimetry.⁴⁰ If the complexes of CTAB–AIBN were formed at interfaces with a lowered activation energy, increasing temperature provides extra energy and the kinetics for AIBN decomposition at interfaces should be accelerated. Increasing thermal energy may not have a significant effect on the interfacial energy at oil–water interfaces, but the diffusion of molecules in



Fig. 1 $log(A_t/A_0)$ as a function of time for AIBN in toluene (bulk), emulsion, and emulsion gel samples at 20, 40, and 60 °C.

Table 1 The decomposition rate constant for AIBN in total ($k_{d,t}$, s^{-1}), in bulk ($k_{d,b}$, s^{-1}), at interface ($k_{d,i}$, s^{-1}), and the rate of polymerization (R_{p} , mol L⁻¹ h⁻¹) measured at 20, 40, and 60 °C

Temperature	Toluene (bulk)		Emulsion			Emulsion gel		
	$k_{\rm d,b} imes 10^{-6}$	Rp	$k_{\rm d,t} imes 10^{-6}$	$k_{ m d,i} imes 10^{-6}$	Rp	$k_{\rm d,t} imes 10^{-6}$	$k_{ m d,i} imes 10^{-6}$	Rp
20 °C	~0	N/A	0.31	0.31	0.08	0.61	0.61	0.28
40 °C	0.19	N/A	0.86	0.67	0.76	2.78	2.59	3.62
60 °C	21.97	N/A	42.58	20.61	4.42	37.11	15.03	6.08

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emulsions increases with temperature.⁴⁵ As temperature elevates, AIBN molecules are expected to migrate to the surfactant-stabilized oil-water interfaces faster, at which they decompose with the assistance of additional interfacial energy. With increased temperature, both interfacial energy and complexation mechanisms are expected to have a greater catalytic effect on the AIBN decomposition at interfaces than in bulk. Indeed, the increment of the $k_{d,i}$ in emulsions (3.6 × 10^{-7} s⁻¹) from 20 to 40 °C is greater than that of the $k_{d,b}$ (1.9 × 10^{-7} s⁻¹) in bulk phases. At 60 °C, as the thermal energy overcomes the activation energy for AIBN decomposition, AIBN can decompose in an emulsion system regardless of their locations. Therefore, the difference between $k_{d,b}$ and $k_{d,i}$ is minimized.

To further investigate the possible mechanisms, fumed silica was added to the emulsion systems. It is worth noting that the fumed silica particles used in this study are of high purity without any potential catalytic elements and impurities.46 On one hand, fumed silica surfaces are negatively charged which adsorb cationic surfactant CTAB readily in emulsions.47 By introducing fumed silica to a similar emulsion system, the amount of CTAB surfactants at oil-water interfaces was reduced by 50% for an emulsion formed with a nonpolar monomer and 10% for that with a polar monomer.38,39 The adsorbed CTAB surfactants on fumed silica surfaces form densely packed aggregates,⁴⁷ which are unlikely available for complexation with AIBN limited by diffusion as illustrated in Fig. 2. Since the molar ratio of surfactant and AIBN used in this study is 1:1, smaller than the proposed complex ratio of 3:1, a decrease in the amount of CTAB surfactants at interfaces should reduce the probability of complex formation, consequently lowering the rate constants for the AIBN decomposition. This expectation is contradictory to the experimental results where the AIBN decomposition is more pronounced in the presence of fumed silica. The $k_{d,i}$ for the silica-containing emulsion gels are 2.0 times and 3.9 times greater than that without silica at 20 and 40 °C, respectively. The decomposition of AIBN in emulsion systems is unlikely due to the complexation with surfactants.

On the other hand, fumed silica particles can form a three-dimensional network and bridge the dispersed oil phases in emulsions,³⁹ leading to gelation in emulsions. It has been shown that high viscosity promotes the AIBN at interfaces.^{32,38,39} decomposition The enhanced decomposition of AIBN in emulsion gels is likely due to the gelation from fumed silica. In the emulsion gels at 60 °C, the decomposition of AIBN contributed from the interfaces was smaller than that in the bulk phases. The rheological properties of the emulsion gels provide a clue for this phenomenon. As seen in Fig. 3, the loss factors $(\tan \delta)$, the ratios of the storage modulus and the loss modulus, for the emulsion gels are smaller than 1, indicating that these samples behaved elastically under shear.⁴⁸ The $\tan \delta$ of the emulsion gels at 20 and 40 °C are almost identical. The emulsion gels at 60 °C had a greater value of the tan δ , suggesting that the gel structures were weakened at 60 °C. As the temperature increases, the aggregate of fumed silica and the surfactant-stabilized oil-water interfaces in emulsions



Fig. 2 Proposed locations for AIBN decomposition in the silicacontaining emulsion gels at 20 °C.



Fig. 3 Loss factor $\tan \delta$ as a function of frequency for the emulsion gels at 20, 40, and 60 °C.

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may change. The temperature-induced rearrangement of fumed silica particles leads to enhanced gelation with increased temperatures,³⁴ which is contradictory to the observations at 60 °C in this study. At high temperatures, the desorption of surfactants became more pronounced and consequently destabilized oil-water interfaces in emulsion gels.⁴⁸⁻⁵⁰ As a result, the desorption of the surfactants from oil-water interfaces reduces the interfacial viscosity and the corresponding gel strength.⁵¹ It is reasonable to deduce that the destabilized oil-water interfaces reduce the interfacial energy, providing insufficient loci for AIBN to decompose. The interfacial initiation and the catalytic effect of silica became less pronounced as the interface destabilized at high temperatures.

In addition, different surfactants should create different electrostatic environments and diffusion resistance near the oil-water interfaces,⁵² which affects the redox reactions and complex formation. The decomposition of thermal initiators in emulsions was found to be independent of the surfactant type as well as the initiator type.¹³ Therefore, it is not surfactant or their complexes but the nature of oil-water interfaces, such as interfacial energy, contributes to the catalytic decomposition of AIBN at low temperatures. An emulsion system with stable interfaces is critical to interfacial initiation. Further study is needed to determine the detailed mechanism of the catalyzed decomposition of AIBN at oil-water interfaces.

3.2 Polymerization at low temperatures

The radicals generated through interfacial initiation are sufficient to initiate free radical polymerizations at low temperatures. The emulsion and emulsion gel samples were polymerized successfully in a dark environment without inert gas protection at 20, 40, or 60 °C. High monomer–polymer conversions were reached for these samples (Fig. 4a and b). At the early stage of free radical polymerization, the monomer–polymer conversion follows a linear relationship.²⁹ The average rates of polymerization in emulsions and emulsion gels were determined from the slope of the following equation

$$R_{\rm p} = -\frac{{\rm d}[{\rm M}]}{{\rm d}t} = \frac{\Delta[{\rm M}]}{\Delta t}$$

where $\Delta[M]$ is the concentration change of the monomer in the period Δt . As the rate of free radical polymerization is affected by many factors, such as viscosity,^{29,46,53} initiation efficiency,^{44,54} and solvent/impurities,^{39,44,55} the contributed from interfacial initiation cannot be calculated simply. Nonetheless, the rates of polymerization for the emulsion and emulsion gel samples are consistent with the total decomposition rate constants of AIBN. The $R_{\rm p}$ for emulsions and emulsion gels increased with temperature and the polymerization in emulsion gels is faster than that in emulsions. It is known that high viscosity usually slows polymerization by increasing the radical termination rate. In this study, as the decomposition of AIBN in emulsion gels is much faster than that in emulsion, considering a relatively small initiation efficiency (<0.4) for a slow-diffusion system,⁵⁴ more active radicals are still present in the emulsion gels than in the emulsions at the same temperature. High radical flux initiates more radical chains in a polymerization system, therefore the monomers are consumed more rapidly. The polymerizations conducted at 20 °C are relatively slow, which took 2-3 days to reach high conversions. The rates of polymerization in emulsion gels at 40 °C are comparable with that in emulsions at 60 °C, about 3-4 hours to reach 90% conversion. By utilization of interfacial initiation, radical polymerizations can be conducted at low temperatures in emulsion gels with a comparable polymerization rate as these conducted at 60 °C.

Due to the exothermic nature of chain propagation,^{53,56} the polymerizations in emulsions and emulsion gels at 60 °C are more intense than their low-temperature counterparts. The surface temperature of the glass vials increases rapidly during polymerization, especially for large-scale production (20 g or above). The heat generated from the chain propagation decomposed more AIBN molecules in the systems, which accelerated the overall polymerization and increased the viscosity dramatically. It is also known as the Trommsdorff effect.⁵³ As a result, the nitrogen molecules



Fig. 4 The monomer-polymer conversion (% by mass) in (a) emulsion and (b) emulsion gel samples at different temperatures. (c) The appearance of the as-synthesized monoliths from the emulsion (left) and the emulsion gels (right) at 20 °C.

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released by AIBN decomposition became more difficult to escape and consequently were trapped in the as-synthesized ABS monoliths. The monoliths polymerized at 60 °C have many voids (Fig. S1†), and their measured densities, 0.67–0.74 g cm⁻³, are much smaller than the density of pure ABS (1.04 g cm⁻³).^{57,58} These voids originated from the fast decomposition of AIBN triggered by the Trommsdorff effect.

In comparison, the initiation and the subsequent polymerization conducted at 20 and 40 °C are steadier. The low-temperature environment dissipates heat generated during polymerization more efficiently. The surface temperatures for the glass vials remain almost the same during the polymerization. The monoliths polymerized in the emulsions and emulsion gels at low temperatures, as shown in Fig. 4c, are uniform in appearance. The densities of these monoliths synthesized at low temperatures are found to be 1.02-1.03 g cm⁻³ which are close to that of neat ABS materials. As the Trommsdorff effect was minimized at low temperatures, the nitrogen gas byproducts were easier to diffuse out of the systems. In addition to energy efficiency, conducting polymerization at low temperatures also improved the safety of the polymerizations and obtained products with a more uniform appearance. Compared with high-temperature polymerizations, conducting polymerization at low temperatures may not require cooling devices, and the production is as efficient as those conducted at 60 °C.

3.3 Characterization and properties

The weight-average molar mass (M_w) and the dispersity (D) for the extracted ABS copolymers are listed in Table 2. Compared with bulk or solution polymerization, chain propagation in emulsion systems takes place more separately in these surfactant-stabilized particles, which produces polymers with higher molecular masses.⁴³ At 60 °C, an $M_{\rm w}$ of 74 and 96 kg mol⁻¹ were obtained for the ABS copolymers synthesized in emulsions and emulsion gels, respectively. High temperatures facilitated the diffusion of radicals and propagating chains, which facilitated intermolecular chain transfer or bimolecular termination at the high conversion (>90%) stage.^{43,59} It agrees with the relatively broad molar mass distribution obtained for 60 °C. It is difficult to synthesize polymers with an ultra-high molar mass and narrow molar mass distribution through free radical polymerization at high temperatures. Additionally, the molar mass distribution for the ABS copolymers at 60 °C is bimodal (Fig. 5), which is an indicator of the presence of the Trommsdorff effect.53

Table 2 The weight-average molar masses (kg mol^{-1}) and the dispersity (D) of the extracted ABS copolymers

	Emulsion		Emulsion gel	gel
Temperature	$M_{ m w}$	Đ	$\overline{M_{ m w}}$	Đ
20 °C	1256	1.7	1416	1.6
40 °C	274	1.9	693	2.1
60 °C	74	4.1	96	2.8



Fig. 5 GPC traces for the extracted ABS copolymers synthesized in the emulsions (E) and the emulsion gels (EG) at 20, 40, and 60 $^\circ$ C.

The $M_{\rm w}$ of the ABS copolymers increases as the polymerization temperature decreases. The molecular masses obtained from the ABS copolymers synthesized at low temperatures are very large, up to 1416 kg mol⁻¹. At low temperatures, the initiation of AIBN was relatively slow, and a small number of propagating chains were initiated. The monomers available for each propagating chain to react with are more abundant than those with high radical flux. In addition, the low temperature facilitates exothermic chain propagation and minimizes chain diffusions. The polymer chains grow rapidly and reach an ultra-high molar mass before termination occurs. Surprisingly, the molar mass distribution became narrower as the polymerization temperature decreased. As the polymer chain grows very long at low temperatures, the active chain ends are easily trapped within the polymer coils and minimize the probability of encountering another propagating chain ends or other active transfer species.⁶⁰ As a result, the bimolecular termination and chain transfer at low temperatures were limited even at a high conversion stage. polymerization temperature As the decreases, the distribution becomes unimodal, suggesting a mild and steady polymerization. The ABS copolymers with an ultrahigh molar mass and narrow molecular distribution were synthesized at low temperatures.

The chemical compositions of the extracted ABS copolymers were analyzed using FTIR and NMR. The spectra can be found in the ESI.† The spectra for the ABS copolymers synthesized in emulsions and emulsion gels at different temperatures are very similar to each other, and also in line with those in literature.^{61–64} The chemical compositions of the as-synthesized ABS copolymers were determined from the NMR spectra.⁶³ The as-synthesized ABS copolymers consist of acrylonitrile/butadiene/styrene segments 38/6/56 by mass, which is close to the original compositions for the monomers in the emulsion or emulsion gel systems. It suggests that the polymerization temperature has little effect on the chemical structures and compositions of the ABS copolymers synthesized in this study.



Fig. 6 DSC thermograms for the extracted ABS copolymers synthesized in the emulsions (E) and the emulsion gels (EG) at 20, 40, and 60 $^\circ C.$

The glass transitions for the extracted ABS copolymers are plotted in Fig. 6. The ABS copolymers synthesized at 60 °C have glass transition temperatures (T_g) at 96–105 °C which is consistent with the ABS copolymers with similar molecular masses.^{65,66} The T_g of the ABS copolymers increased slowly as the polymerization temperature decreased. As the chemical compositions for the as-synthesized ABS copolymers are similar, the increase in the $T_{\rm g}$ is likely due to their molecular masses. According to the Fox-Flory equation, the $T_{\rm g}$ of a polymer depends on the free volume created by the mobile chain ends.67 The free-volume effect minimizes in the polymers with an ultra-high molecular mass. The $T_{\rm g}$ s saturate at a higher temperature, 115 °C, for the ABS copolymers with a molar mass greater than 1200 kg mol^{-1} (synthesized at 20 °C).

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The morphologies of the as-synthesized monoliths are shown in Fig. 7. The ABS materials synthesized at 20 °C are spherical particles with a diameter of 231 ± 65 (emulsions) and 190 ± 36 nm (emulsion gels). Similar spherical morphologies were obtained for these synthesized at 40 °C with a smaller size, e.g., 118 ± 14 and 91 ± 11 nm from the polymerization of emulsions and emulsion gels, respectively. The differences in the particle sizes originated from the molecular masses of the incorporated ABS copolymers. The ABS copolymers synthesized at lower temperatures have a longer polymer chain which occupied a larger volume and consequently produced larger particles. At 60 °C, the polymerization of the emulsions produced spherical particles with a diameter of 123 ± 49 nm while that of the emulsion gels formed porous structures without a welldefined particle shape. This morphology is consistent with the rheological results where the oil-water interfaces in emulsion gels were destabilized at 60 °C. The voids or pores observed in the ABS monoliths synthesized from emulsion gels at 60 °C likely originated from the trapped nitrogen gas during polymerization.

The mechanical properties of the as-synthesized monoliths were tested using nanoindentation. The presence of silica particles created heterogeneous surfaces in the monoliths, which complicates the surface mechanical analysis. To avoid the effects of silica, only these monoliths synthesized from emulsions were tested. As the chemical compositions and microstructures for these monoliths from emulsions are very similar, the major difference among these monoliths is the molar mass of the ABS copolymers, which varied from 74 kg mol⁻¹ (synthesized at 60 °C) to 1256 kg mol⁻¹ (synthesized at 20 °C). Theoretically, the presence of small molecules (surfactants) and the interfaces/voids between polymer



Fig. 7 Scanning electron micrographs of the ABS monolith synthesized from emulsions (E) or emulsion gels (EG) at 20, 40 or, 60 °C. The scale bar represents 1 μm.

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particles in the monoliths should alter, most likely deteriorate, the overall mechanical strength of the materials. Therefore, the mechanical properties of the as-synthesized monoliths are not equal to that of the synthesized ABS copolymers. The tests on these monoliths are still meaningful as they provide a first insight into the mechanical properties of these ABS copolymer materials synthesized at low temperatures.

As seen in Fig. 8a, the Young's moduli for the monoliths synthesized at different temperatures are very close to each other, which are 4.3-4.5 GPa. The same trend was also observed for Vickers hardness and elastic deformation work (Fig. 8b and c). The ultra-high molecular masses of the ABS copolymers have little effect on these mechanical properties. The measured plastic deformation work as well as the total deformation work (elastic work + plastic work) are greater for these monoliths synthesized at lower temperatures. The greater total deformation work suggested that the monoliths are capable of absorbing more energies upon deformation, which is an indicator of fracture toughness.⁶⁸ The brittleness of the ABS monoliths was estimated using the ratio of elastic work and plastic work.^{68,69} The ratios calculated for the ABS monoliths synthesized at 20, 40, and 60 °C are 0.37, 0.43, and 0.50, respectively. It suggested that the ABS monoliths synthesized at lower temperatures are less brittle or more ductile.⁶⁸

Traditionally, ductile ABS materials were synthesized by incorporating more butadiene segments into the copolymers, which deteriorates the modulus and hardness of the materials.^{70,71} As the chemical compositions of the assynthesized ABS copolymers are almost identical, the increment in ductility is unlikely due to the butadiene segments. Similar to other polymers with ultra-high molecular masses, long polymer chains can entangle and tie each other to a greater extent, which can transfer energy to the polymer backbone more efficiently.72 As more energy was absorbed upon deformation, the ABS monoliths with higher molecular masses were more ductile. The interface-initiated room-temperature polymerizations provide an alternative approach to enhance the ductility of ABS materials without compromising their modulus and surface hardness. The narrower molar mass distribution for the ABS copolymers synthesized at low temperatures is also desirable for highperformance applications because the mechanical properties are more predictable and reliable.⁷³

4. Conclusion

ABS copolymers can be synthesized at low temperatures (20 or 40 °C) efficiently. The decomposition of AIBN contributed from the oil-water interfaces was measured for the first time. At low temperatures, the decomposition of AIBN at interfaces or interfacial initiation dominates the initiation in emulsion systems. Interfacial initiation depends on the temperature and the stability of the interfaces. The polymerization conducted through interfacial initiation meets the requirements for green chemistry in the following aspects.¹⁶ Energy efficiency: the lowered polymerization temperature saved tremendous energy for polymer production. It reduces greenhouse gas and pollutant emissions and improves the carbon footprint of polymer industries. Waste prevention and atom economy: the monomer conversions for interface-initiated room-temperature emulsion polymerizations are greater than 90%. They are the same as these emulsion polymerizations conducted at high temperatures.⁷⁴ Compared with traditional techniques, roomtemperature emulsion polymerization leaves no additional chemical waste to treat or clean up afterward. Accident prevention: the low-temperature approach can be conducted safely in a mild and steady condition. Additional cooling devices may be unnecessary if polymerization is conducted at low temperatures. It makes the emulsion polymerization process safer and simpler.

In addition to being eco-friendly, low-temperature polymerization also produced ABS with ultra-high molecular masses and narrow molar mass distribution. Compared with the ABS materials obtained at 60 °C, those from the low-temperature polymerization showed higher $T_{\rm g}$, and better ductility without compromising their modulus and surface hardness. Conducting low-temperature polymerization through interfacial initiation is an efficient eco-friendly technique to produce high-performance polymer materials.

Associated content

The picture for the monoliths synthesized at 60 °C, the FTIR spectra, and the NMR spectra were available in the ESI.†

Data availability

Data will be made available on request.

Author contributions

The manuscript was written through the contributions of all authors. All authors have approved the final version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

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