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# Thermodynamic driving forces for autoreduction of Cu sites in the zeolite SSZ-13

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Cu-exchanged zeolites are widely used redox catalysts and the oxidation state of Cu is crucial in understanding their performance over a wide range of applications. Interestingly, a fraction of Cu sites in zeolites is reported to reduce at high temperatures in the absence of a reducing agent and as of today a detailed understanding of this process is still missing. In this contribution, we use first principles-based phase diagrams to explore thermodynamic driving forces for the autoreduction of Cu sites in the zeolite SSZ-13. We find that mainly monovalent Cu(I)-OH sites anchored at well-separated Al atoms and to a lesser degree di-hydroxyl Cu dimers drive the autoreduction of Cu sites in the zeolite SSZ-13. Using these insights, we can reproduce experimental trends in autoreduction reported in the literature. This work gives detailed insights into the autoreduction of Cu sites in SSZ-13 and demonstrates that the nature of Cu sites in the zeolite SSZ-13 depends on the exact conditions the material is exposed to. Optimizing these reaction conditions might allow to improve the performance of Cu-exchanged zeolites over a wide range of applications.

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## Introduction

Metal-exchanged zeolites are often used catalysts for redox reactions<sup>1–6</sup> and in particular their Cu-exchanged forms have drawn significant attention over the past years.<sup>7–15</sup> Their main applications include the selective catalytic reduction of nitrous oxides in the presence of NH<sub>3</sub><sup>3,4,8,16–19</sup> and the partial oxidation of methane<sup>7,9,10,20–23</sup> and other saturated alkanes. One of the key features in this context is the ability of the Cu site to change its oxidation state. Under oxidizing conditions, Cu can be found in an oxidation state of +2, while oxidation states of +1 or in rare cases 0 have been reported<sup>24</sup> under reducing conditions.

Typically, the change in oxidation state is associated with the presence of an oxidizing or reducing agent. However, in the literature changes in the oxidation state of Cu sites have also been reported for the exposure of Cu-exchanged zeolites to inert conditions during the so-called Cu-autoreduction.<sup>25–34</sup> Here, a fraction of Cu sites changes their oxidation state from +2 to +1 upon exposure of the system to inert He at high temperature. So far, several factors, such as the zeolite framework structure, the Si/Al ratio, and the Cu loading, have been shown to impact the autoreduction of Cu sites.<sup>25</sup> Additionally, a phenomenological model for the reaction mechanism, which

includes (i) the loss of H<sub>2</sub>O from 2 Cu(II)-OH sites and (ii) the desorption of  $\frac{1}{2}$  O<sub>2</sub>, has been suggested.<sup>25,32</sup> So far, these insights are purely experimentally based and a theoretical model for these processes is still missing.

A zeolite that has gathered significant attention over the past years is Cu-exchanged SSZ-13, a medium- to high-silica zeolite in the chabazite framework structure. This material has been introduced as the industrial catalyst for the selective catalytic reduction of nitrous oxides<sup>3,4,19,35</sup> and shows potential for the partial oxidation of methane to methanol.<sup>20,21,36–38</sup> For this system, multiple studies have investigated the nature of the active sites and it was initially believed that mainly single Cu atoms in oxidation states of either +1 or +2 exist.<sup>16,24,39–43</sup> More recent work, however, indicates that, depending on the exact encountered conditions, different Cu oxo- and hydroxyl dimers, ranging from Cu<sub>2</sub>O over Cu<sub>2</sub>OH to Cu<sub>2</sub>(OH)<sub>2</sub>, can also be present inside this material.<sup>21,36,37</sup>

Many of these insights have been obtained from first principles calculations-based, complex phase diagrams<sup>16,36,44,45</sup> for the conversion of methane to methanol during the stepwise partial methane oxidation. These phase diagrams rely on the premise that if a system is exposed to specific conditions for an extended period, all chemical reactions are sufficiently fast that the system can reach its thermodynamic ground state at the given conditions. This precondition is met for a stepwise process in partial methane oxidation, where the system stays under the same conditions for hours, but also during autoreduction of Cu sites, where the system is kept at specific conditions for an extended time. Phase diagrams could

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therefore be used to gain detailed insights into changes in active site distribution in Cu-SSZ-13 during autoreduction.

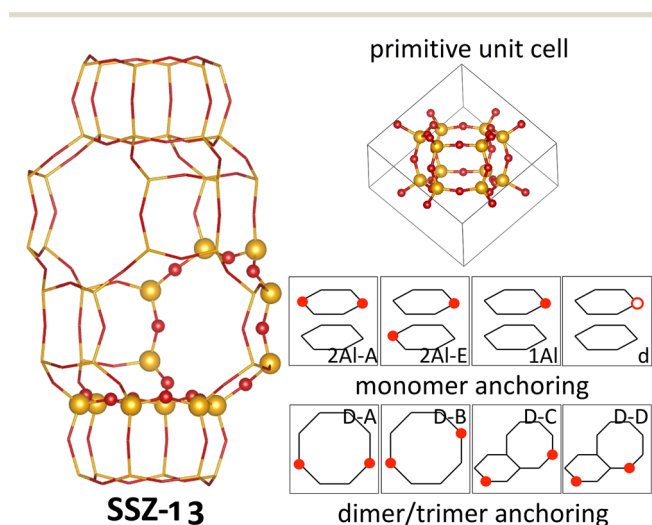
Herein, we use phase diagrams to explore the oxidation state of Cu sites in the zeolite SSZ-13 under various conditions. We explore the impact of Al arrangements for monomer and dimer anchoring on Cu oxidation state and relate these insights to the autoreduction of Cu sites in SSZ-13 observed in experimental measurements.

## Methods

### The system

In this work we focus on SSZ-13, a zeolite in the chabazite framework. The chabazite framework is the zeolite framework with the smallest primitive unit cell, which only contains 12 symmetrically equivalent tetrahedral sites (T-sites) linked by O atoms (see Fig. 1). More specifically, the primitive unit cell consists of double six-O-membered ring units (d6R), which are linked by four-O-membered rings. In between the d6R structures, a pore is formed which is connected to adjacent pores *via* eight-O-membered rings. The small unit cell size and the high symmetry makes the chabazite framework the ideal model system to understand zeolites.<sup>46</sup>

If all T-sites are occupied by Si, the system is chemically inert, but when T sites are occupied by Al, a local negative charge is generated which needs to be compensated by the presence of either a proton, another cation, or a larger, charged cluster. The exact nature of the exchanged cation depends on the anchoring configuration, *i.e.*, the exact configuration of Al atoms surrounding it.



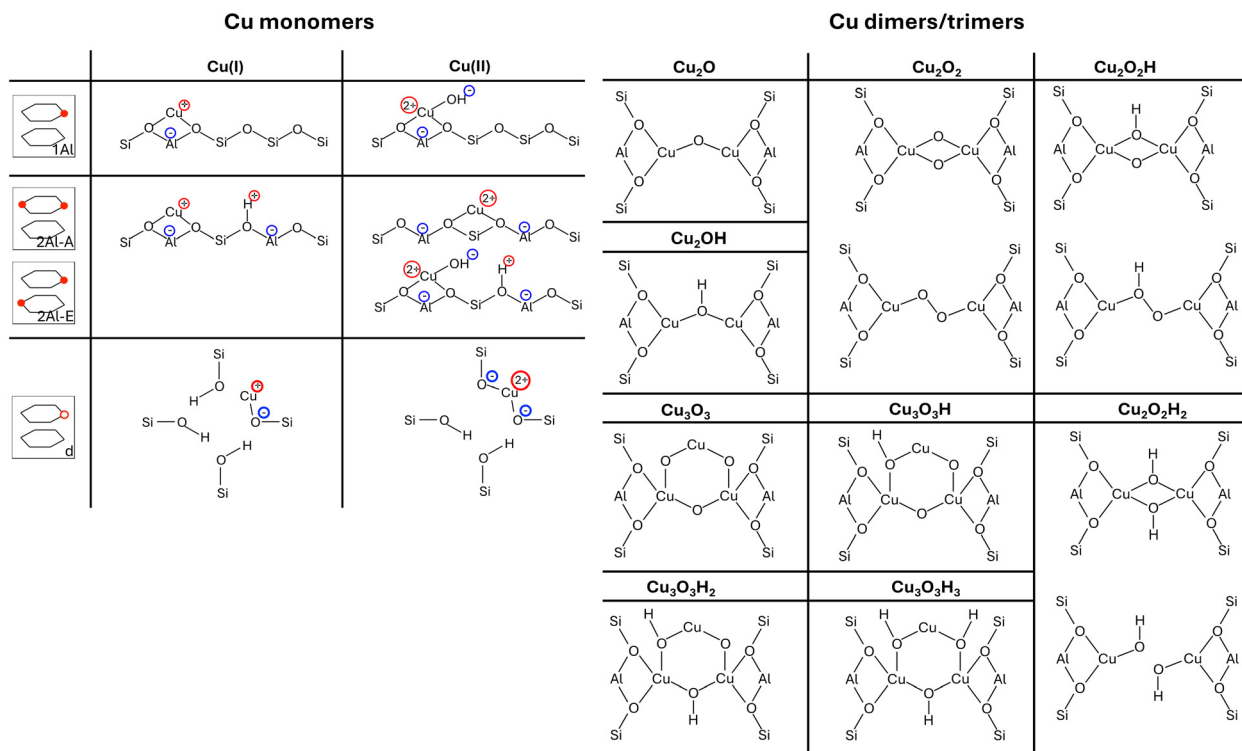
**Fig. 1** Main cavity (left) and primitive unit cell (right top) of the zeolite SSZ-13. In the atomistic representations, Si is shown in yellow, and O is shown in red. In the picture of the main cavity, the six- and eight-O-membered rings are highlighted, while the remaining framework is shown in a stick representation. Schematic representations of anchoring configurations for Cu monomers and dimers/trimers are shown in the bottom right. Here, Si(Al)-O bonds are shown as black lines, with Al positions marked by red circles.

In the analysis in this work, we include four different Cu-monomer anchoring configurations,<sup>36,44,47</sup> namely a configuration in which two Al atoms are on opposite sides of a six-O-membered ring (2Al-A), a configuration in which two Al atoms are positioned in different six-O-membered rings of the same primitive unit cell (2Al-E), and a configuration in which one Al atom is placed in one primitive unit cell (1Al). Additionally, we also include a silanol defect for monomer anchoring (d), which is obtained by removing one Si/Al atom from the framework and saturating the remaining bonds of four O atoms with H. These four monomer anchoring positions describe four different scenarios, namely a scenario in which Al atoms are paired in the same six-O-membered ring (2Al-A), a scenario in which Al atoms are close, but not in the same six-O-membered ring (2Al-E), a scenario in which all Al atoms are well separated (1Al), and a scenario in which framework defects are present in the material (d). All Al configurations are shown in Fig. 1. For these Al configurations, we add monovalent (Cu(I)) and divalent Cu monomers (Cu(II)) with up to six adsorbed water molecules. A schematic representation of the different Cu monomers before hydration is given in Scheme 1. Details on how these sites are generated and structural files for them are given in the literature.<sup>36,42,44</sup> Upon water adsorption, the coordination of Cu to the framework in exchange positions 2Al-A, 2Al-E, and 1Al is reduced upon water adsorption until Cu sites are solvated and form a linear coordination to two water molecules for Cu(I) and a tetrahedral coordination to either four water molecules for 2Al-A and 2Al-E or three water molecules for the 1Al configuration. Additional water molecules form hydrogen bonds either with water coordinated to Cu or to Brønsted acid sites in the unit cell. These trends agree with other structures reported in the literature.<sup>16</sup> For the defect site, Cu remains coordinated to the framework O atoms.

Additionally, we consider four different Al configurations for the anchoring of dimer and trimer structures (D-A through D-D), two of which are in the eight-O-membered ring while the other two bridge a six-O-membered ring and an eight-O-membered ring. All configurations are displayed in Fig. 1. For these dimer exchange sites, we include Cu-oxo and Cu-hydroxyl dimers with one or two oxo or hydroxyl groups. For D-A and D-B, we also include Cu trimers with three oxo or hydroxyl groups. During optimization, different structural motifs were explored for all structures and structural files for these sites can be found in the literature.<sup>36</sup> Furthermore, schematic representations of the different Cu dimers and trimers are shown in Scheme 1.

Throughout this work, we focus on calculations for a single, primitive, 12T chabazite unit cell, which shows a too high Al concentration ( $\text{Si}/\text{Al} = 5$ ) compared to typical experimental measurements ( $12 < \text{Si}/\text{Al} < 20$ ). However, past tests for dimers anchored in the D-C dimer exchange position have shown that these calculations reproduce results for a 24T chabazite unit cell ( $\text{Si}/\text{Al} = 11$ ), within 0.06 eV.<sup>36</sup>





**Scheme 1** Schematic representation of all Cu(I) and Cu(II) monomers before hydration (left) and optimized dimer and trimer structures (right) included in our model. Monomer configurations are displayed for each exchange position, which is displayed schematically similar to Fig. 1. If multiple structures are displayed for the same dimer/trimer stoichiometry, the top structure is found for the D-A and D-B exchange position, while the bottom structure is found for the D-C and D-D exchange positions.

## Phase diagrams

To compare the relative stability of different Cu sites, we will rely on phase diagrams. Here, we calculate the energy of each structure and extrapolate to the finite-temperature Gibbs free energies by adding zero-point energy corrections and vibrational, rotational, and translational entropies. Many of the Cu species have a different stoichiometry and we cannot directly compare their stabilities. We therefore calculate  $\mu_{\text{Cu}_x\text{O}_y\text{H}_z}^{\text{Cu}}$ , the chemical potential of Cu for Cu site  $\text{Cu}_x\text{O}_y\text{H}_z$ , as<sup>36</sup>

$$\mu_{\text{Cu}_x\text{O}_y\text{H}_z}^{\text{Cu}}(T, P^{\text{O}_2}, P^{\text{H}_2\text{O}}) = \left( G^{\text{Cu}_x\text{O}_y\text{H}_z-\text{zeo}}(T) - G^{2\text{H}-\text{zeo}}(T) - \frac{2y-z+2}{4} \mu^{\text{O}_2}(T, P^{\text{O}_2}) - \frac{z-2}{2} \mu^{\text{H}_2\text{O}}(T, P^{\text{H}_2\text{O}}) \right) / x$$

for an Al configuration with two Al atoms in the primitive unit cell (2Al-A, 2Al-E, and all dimer anchoring configurations) and

$$\mu_{\text{Cu}_x\text{O}_y\text{H}_z}^{\text{Cu}}(T, P^{\text{O}_2}, P^{\text{H}_2\text{O}}) = \left( G^{\text{Cu}_x\text{O}_y\text{H}_z-\text{zeo}}(T) - G^{1\text{H}-\text{zeo}}(T) - \frac{2y-z+1}{4} \mu^{\text{O}_2}(T, P^{\text{O}_2}) - \frac{z-1}{2} \mu^{\text{H}_2\text{O}}(T, P^{\text{H}_2\text{O}}) \right) / x$$

for the 1Al configuration, where only one Al atom is present in the unit cell. In these equations,  $G^X$  refers to the Gibbs free energy of structure X, zeo is shorthand notation for the zeolite, which varies for each anchoring configuration (2Al-A, 2Al-E, 1Al, and d), and  $\mu^{\text{O}_2}$  and  $\mu^{\text{H}_2\text{O}}$  refer to the chemical potentials of  $\text{O}_2$  and  $\text{H}_2\text{O}$  in the gas phase, respectively. The gas phase chemical potentials furthermore depend on the gas phase pressures  $P^{\text{O}_2}$

and  $P^{\text{H}_2\text{O}}$ , which are compared to a reference pressure  $P_0$  of one bar. Therefore,  $\mu^{\text{Cu}}$  is dependent on the temperature  $T$  as well as the gas phase pressures of  $\text{O}_2$  ( $P^{\text{O}_2}$ ) and  $\text{H}_2\text{O}$  ( $P^{\text{H}_2\text{O}}$ ).

Overall, the most stable Cu configuration will always be associated with the lowest  $\mu^{\text{Cu}}$ . However, in a realistic zeolite a distribution of different anchoring points will be available. We will therefore study phase diagrams pairing one of the monomer exchange positions 2Al-A, 2Al-E, and 1Al with one dimer exchange position. To understand how the presence of framework defects impacts Cu site preference, we will study phase diagrams which

include one monomer exchange position, the defect position, and, if explicitly mentioned, one dimer/trimer exchange position.

If not stated otherwise, we will include all phase diagrams in the main text. More details considering the construction of phase diagrams can be found in the literature.<sup>36</sup>

## Computational setup

Data used in this work was obtained using the Vienna *Ab Initio* Simulation Package (VASP),<sup>48,49</sup> a plane wave electronic structure



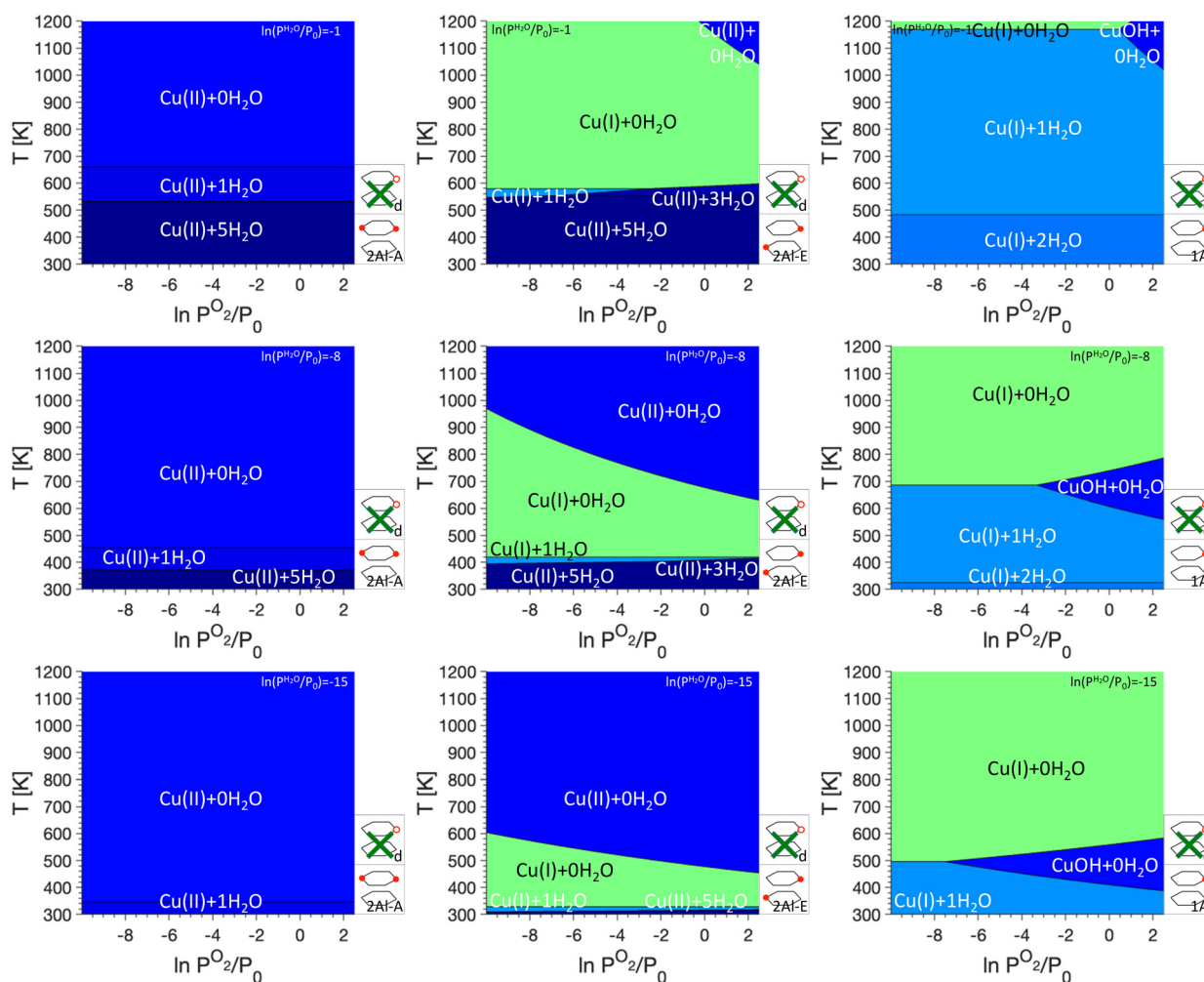
code using PAW pseudopotentials.<sup>50,51</sup> In all calculations we focused on modeling only the  $\Gamma$  point and we relied on a unit cell structure published in the literature<sup>52</sup> and set unit cell volume to 830 Å. All structures were first optimized using the spin-polarized PBE-TS functional.<sup>53,54</sup> For optimized structures, energies were calculated using RPA.<sup>55,56</sup> Gibbs free energies were extrapolated to finite temperature based on the harmonic approximation from vibrational frequencies calculated at the PBE-TS level using Thermo.Pl.<sup>57</sup> In this approach we assume that molecules lose all their translational entropy upon adsorption, but it has been suggested that water might only lose a fraction of translational entropy when binding to Cu in zeolites.<sup>16,58</sup> Therefore, we expect that our approach underestimates the temperature for dehydration of Cu monomers. More details on the computational setup can be found in the literature.<sup>36</sup>

## Results and discussion

The goal of this work is to use phase diagrams to understand changes in the oxidation state of Cu sites under different

conditions, such as Cu autoreduction. As discussed in the Methods section, the relative stability of the Cu sites depends on the temperature  $T$  and the gas phase pressures of  $O_2$  ( $P^{O_2}$ ) and  $H_2O$  ( $P^{H_2O}$ ). Here, we will show  $P^{O_2}/T$  phase diagrams, but for each studied monomer/dimer/defect anchoring combination, we will show phase diagrams for three different water pressures, namely  $\ln(P^{H_2O}/P_0)$  values of  $-1$ ,  $-8$ , and  $-15$ . These three scenarios correspond to exposure to water vapor ( $\ln(P^{H_2O}/P_0) = -1$ ), an atmospheric water pressure ( $\ln(P^{H_2O}/P_0) = -8$ ), and very low water pressure as could be encountered in a vacuum or helium atmosphere ( $\ln(P^{H_2O}/P_0) = -15$ ).

We begin our analysis with the oxidation states for phase diagrams where only Cu monomers are included (see Fig. 2). We immediately see that for 2Al-A, only Cu(II) sites are stabilized. Depending on the water pressure and temperature, Cu(II) coordinated to either five, one, or no water molecules are present, with a higher water coordination found at higher  $P^{H_2O}$  values and lower temperatures. For 2Al-E, we find that at low and high temperatures Cu(II) coordinated to five, three, or no



**Fig. 2**  $P^{O_2}/T$  phase diagrams for monomers anchored in 2Al-A (left), 2Al-E (middle), and 1Al at different  $P^{H_2O}$  values. Regions of uniform color correspond to the site indicated by the label. Schematics of the included anchoring configurations are shown in the bottom right, and green crosses indicate anchoring configurations that are not included.  $P^{H_2O}$  values are given in the legend on the top right.



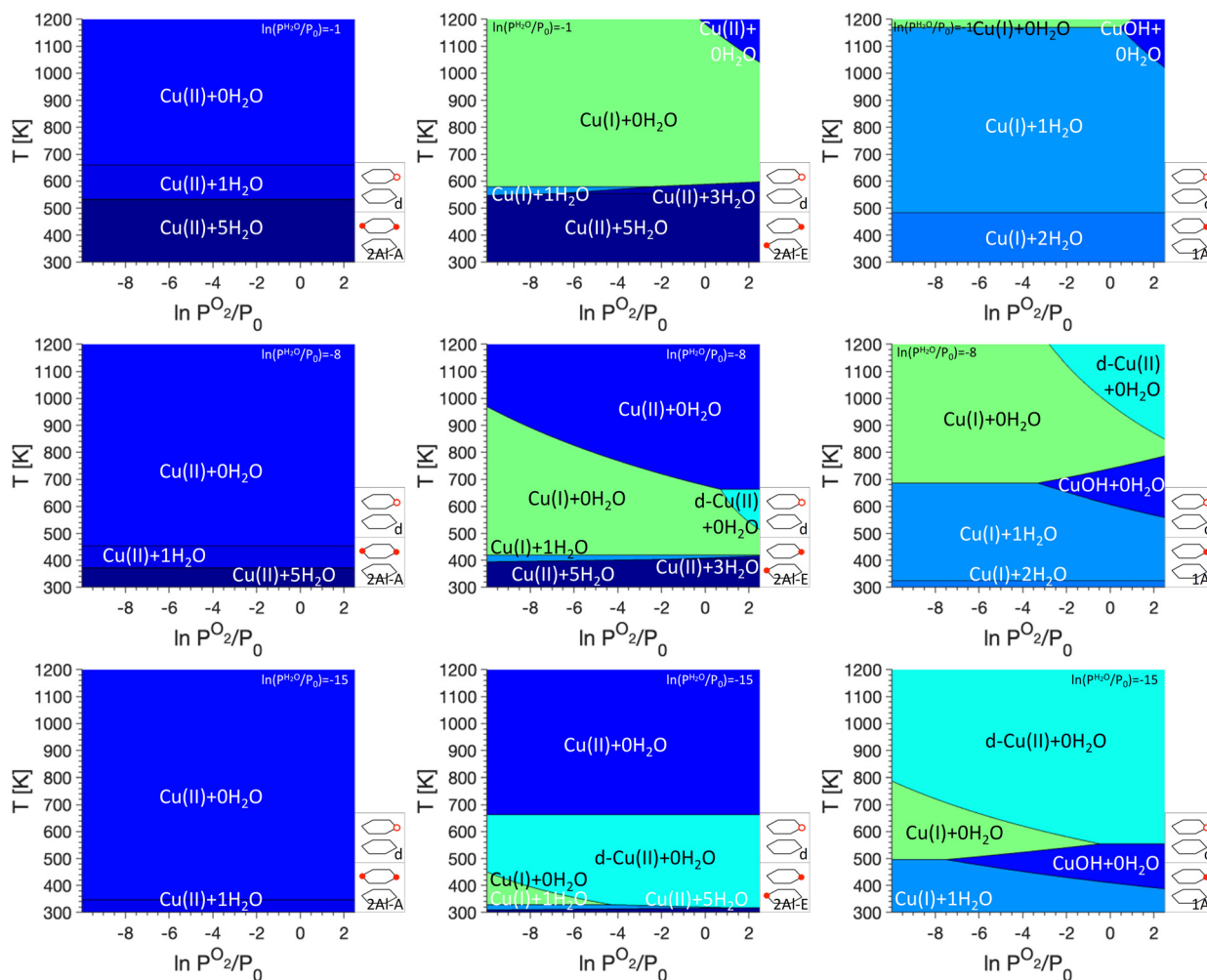


range, however, Cu(I) coordinated to no H<sub>2</sub>O molecules are preferred. The temperature range where Cu(I) is found changes quite significantly between different H<sub>2</sub>O and O<sub>2</sub> pressures. It can range between ~600 K and 1000 K for  $\ln(P^{\text{H}_2\text{O}}/P_0) = -1$  to ~325 K to ~475 K for  $\ln(P^{\text{H}_2\text{O}}/P_0) = -15$ . We attribute the formation of Cu(I) at intermediate temperatures to the presence of only one Al atom per six ring. For the formation of Cu(II), the charges located at the framework and the Cu atom are separated, while the formation of Cu(I) requires the addition of a Brønsted proton. At intermediate temperatures and high  $P^{\text{H}_2\text{O}}$ , this will lead to the formation of Cu(I), while entropic penalties for the addition of a Brønsted proton will stabilize Cu(II) at higher temperatures. For 1Al, mainly Cu(I) coordinated to two, one or no H<sub>2</sub>O is preferred and only at high  $P^{\text{O}_2}$ , a small region of Cu(II)OH is preferred. This region shifts with  $P^{\text{H}_2\text{O}}$  and can be centered around 1100 K for  $\ln(P^{\text{H}_2\text{O}}/P_0) = -1$  or around 500 K for  $\ln(P^{\text{H}_2\text{O}}/P_0) = -15$ .

The introduction of framework defects by including monomer exchange positions and silanol defects for Cu anchoring in our model (see Fig. 3). This does not change

the preferred Cu sites for 2Al-A. For 2Al-E and 1Al, however, the preferred sites change significantly, specifically at low  $P^{\text{H}_2\text{O}}$ . In both cases, a preference of Cu(II) that is not coordinated to water and anchored at a defect, is first seen at  $\ln(P^{\text{H}_2\text{O}}/P_0) = -8$  and  $\ln(P^{\text{O}_2}/P_0) > 0$  for 2Al-E and  $\ln(P^{\text{O}_2}/P_0) > -4$  for 1Al, respectively. For 2Al-E, defect anchored Cu is found at intermediate temperatures of ~600 K, while for 1Al, this Cu species is preferred at temperatures above 800 K. For  $\ln(P^{\text{H}_2\text{O}}/P_0) = -15$ , defect anchored Cu makes up a significant portion of the phase diagrams and for both anchoring configurations, almost the entire region where Cu(I) + 0H<sub>2</sub>O is preferred is replaced by a defect anchored Cu species.

Next, we add dimer anchoring configurations in our analysis (Fig. 4). For 2Al-A, a small region in the phase diagrams shows preference for either Cu<sub>2</sub>O<sub>2</sub>H<sub>2</sub> (D-A and D-B) or Cu<sub>2</sub>OH (D-A), which shifts to lower temperatures for lower  $P^{\text{H}_2\text{O}}$ . For D-C and D-D, dimers are never most stable. For 2Al-E, regions where dimers are preferred increase significantly and at high  $P^{\text{H}_2\text{O}}$  Cu(II) coordinated to five H<sub>2</sub>O is preferred at low temperature, while at lower  $P^{\text{H}_2\text{O}}$  Cu(II) without adsorbed water is found at



**Fig. 3**  $P^{\text{O}_2}/T$  phase diagrams for monomers anchored in 2Al-A (left), 2Al-E (middle), and 1Al with defects included at different  $P^{\text{H}_2\text{O}}$  values. Regions of uniform color correspond to the site indicated by the label. Schematics of the included anchoring configurations are shown in the bottom right and  $P^{\text{H}_2\text{O}}$  values are given in the legend on the top right.

high temperatures. For 1Al, the preference for Cu monomers almost completely disappears for D-A and D-B, while Cu(I) coordinated to one or two water molecules is present for D-C and D-D. Again, a preference for Cu(II) anchored to defects is seen at low  $P_{\text{H}_2\text{O}}$ . Besides the Cu dimers discussed in the literature, our phase diagrams also show a preference for  $\text{Cu}_2\text{O}$  dimers at low  $P_{\text{H}_2\text{O}}$  for monomer anchoring configurations 1Al and to a lesser degree 2Al-E and dimer anchoring configurations D-B and D-D.

In this contribution we aim to understand how the charge state of Cu sites can change with different conditions. We want to assign a charge state to each of the preferred sites encountered in the phase diagrams. For Cu monomers, the situation is straightforward, Cu(I) is charged +1, while Cu(II) has a charge state of +2. Also, for the  $\text{Cu}_2\text{O}_2\text{H}_2$  sites a charge state of +2 can directly be assigned to each Cu atom. However, assigning an unambiguous charge state to Cu bound in  $\text{Cu}_2\text{OH}$  sites is more complicated. This site is

anchored at two Al atoms and contains one OH group. Therefore, the two Cu atoms should have a combined charge state of +3 or +1.5 per Cu atom. However, it is not clear if the charge is equally distributed over both Cu atoms or if one of the Cu atoms has a charge state of +2, while the other Cu has a charge state of +1. To resolve this problem, we study the charge distribution of the lowest unoccupied molecular orbital (LUMO) of the  $\text{Cu}_2\text{OH}$  sites placed in all four dimer anchoring configurations (see Fig. 5). Analyzing these orbitals reveals that for D-A and D-B, the LUMO of the  $\text{Cu}_2\text{OH}$  is delocalized over both Cu atoms. Here, it is not possible to unambiguously assign an integer charge state to each Cu atom and we will identify a  $[\text{Cu}_2\text{OH}]^{2+}$  site. For D-C and D-D the situation is different. Here, the LUMO is almost exclusively located at the Cu atom in the six ring and therefore there is one Cu(II) atom located in the six-O-membered ring and one Cu(I) atom in the eight-O-membered ring.

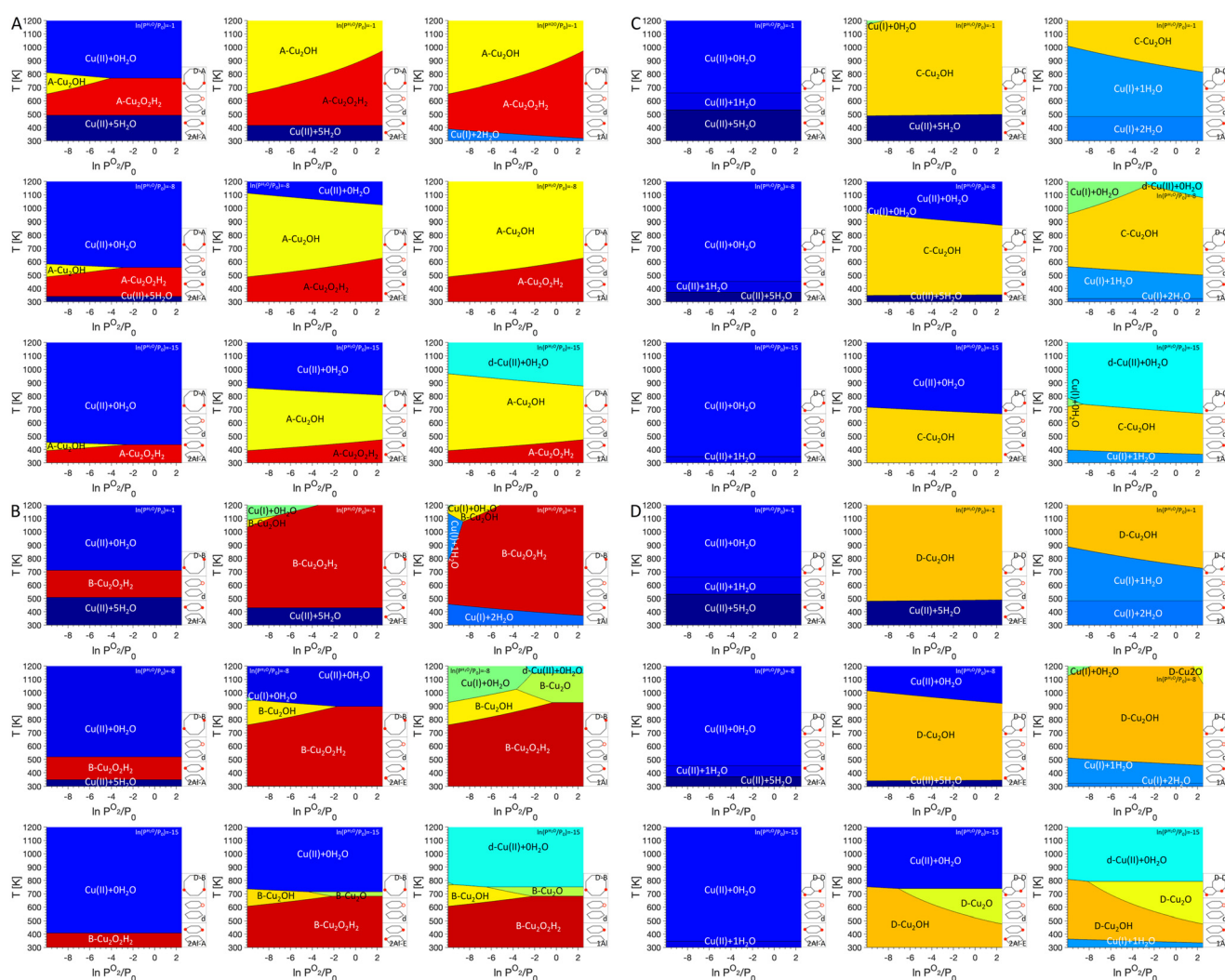


Fig. 4  $P_{\text{O}_2}/T$  phase diagrams for monomers anchored in 2Al-A (left), 2Al-E (middle), and 1Al and dimer/trimer anchoring configurations D-A (A), D-B (B), D-C (C), and D-D (D), with defects included at different  $P_{\text{H}_2\text{O}}$  values. Regions of uniform color correspond to the site indicated by the label. Schematics of the included anchoring configurations are shown in the bottom right and  $P_{\text{H}_2\text{O}}$  values are given in the legend on the top right.



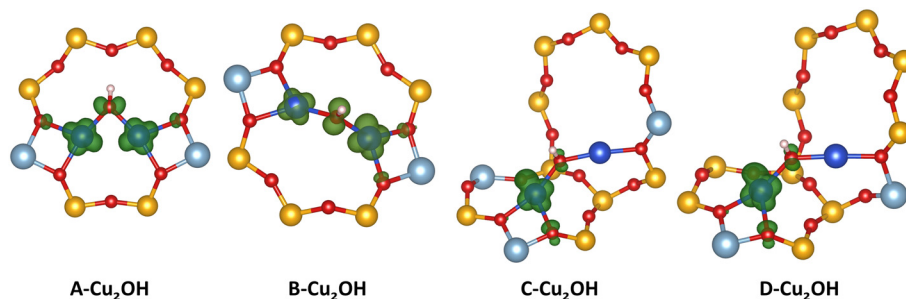


Fig. 5 LUMO orbitals of  $\text{Cu}_2\text{OH}$  sites in different dimer anchoring configurations. The LUMO charge isosurfaces are shown in green, Cu atoms in blue, Si atoms in yellow, O atoms in red, H atoms in white, and Al atoms in grey.

Using this information, it is now possible to understand the changes in charge state based on the encountered conditions. Here, we will assume that the material is activated at 700 K for  $\ln(P^{\text{O}_2}/P_0) = 0$  and  $\ln(P^{\text{H}_2\text{O}}/P_0) = -8$ . Subsequently, autoreduction is started at the same temperature but with a reduction in  $P^{\text{O}_2}$  and  $P^{\text{H}_2\text{O}}$  to  $\ln(P^{\text{O}_2}/P_0) = -7$  and  $\ln(P^{\text{H}_2\text{O}}/P_0) = -15$ . Such a reduction in pressure is consistent with exposing the system to a vacuum or a He atmosphere. Using these assumptions, we find that during activation, all monomers are in the +2 charge state, while dimers D-A, D-C, and D-D are occupied by  $\text{Cu}_2\text{OH}$  sites and  $\text{Cu}_2\text{O}_2\text{H}_2$  is formed for D-B. Upon exposure to autoreduction conditions, only the 1Al monomers change charge state to +1, while the dimers anchored in D-B get reduced to  $\text{Cu}_2\text{OH}$ . If defects are present in the material, Cu will migrate from the 1Al exchange position to the silanol defect and remain in the charge state +2.

Using these assumptions, we can now study several examples of hypothetical Cu-exchanged SSZ-13 samples (see Fig. 6). For all examples, we assume a distribution of exchange positions is available for Cu. Examples 1 and 2 are based on differences in high Si/Al ratio zeolites, which are synthesized with or without framework defects,<sup>44,59</sup> examples 3 and 4 explore the impact of Al pairing for SSZ-13 samples with Si/Al = 15,<sup>60–62</sup> while example 5 focuses on a zeolite with Si/Al = 8. We constructed these examples using the following assumptions: (i) the number of unit cells that contain two Al atoms will be lower at higher Si/Al ratios, (ii) if Al pairs in one six ring exist, they correspond to the 2Al-A configuration,<sup>62,63</sup> and (iii) there is maximum ion exchange and there must be enough Al atoms in the framework to host all Cu during ion exchange (Cu/Al = 0.5). The distributions were then constructed in a way to match available experimental data.

In example 1 we focus on a SSZ-13 zeolite with a Si/Al ratio of 70 without defects. For such a zeolite, we assume that all Al is well separated and only the 1Al anchoring configuration and no dimer anchoring configurations are present. Here, initially all Cu is in a +2 charge state during activation, and all Cu is reduced with the drop in pressure. In example 2, we choose a zeolite similar to example 1 but include enough framework defects to host all Cu atoms. Now, no Cu can be reduced when  $P^{\text{O}_2}$  and  $P^{\text{H}_2\text{O}}$  are lowered, but Cu–OH migrates

from extraframework positions to defect sites and loses the hydroxyl group. In example 3, we focus on a zeolite sample with a Si/Al ratio of 15. Here, we assume that some Al is paired in the six ring and 20% of all Cu can be exchanged in the 2Al-A configuration. Another 20% of Cu can be exchanged with Al atoms that are close, but not paired in the same six ring, which in our model corresponds to ion exchange in 2Al-E. Each dimer exchange position is present and can exchange

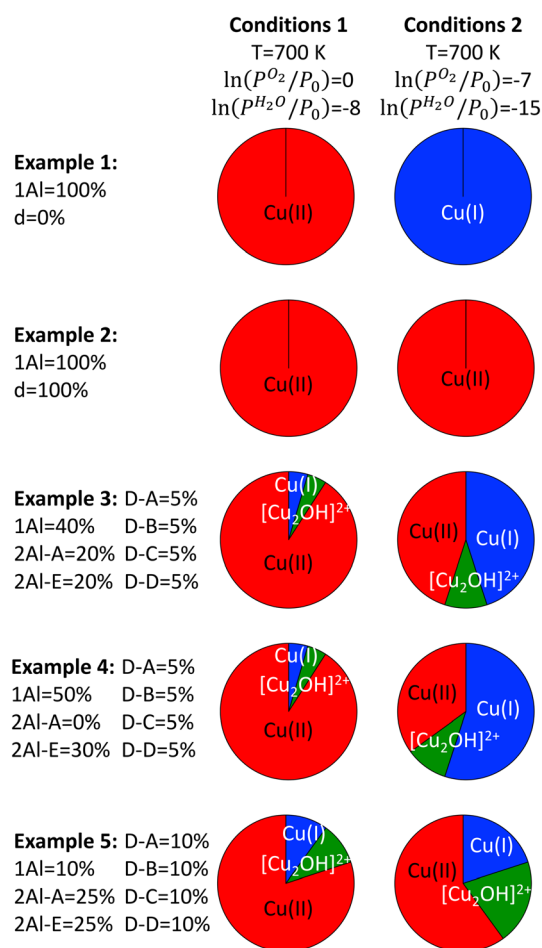


Fig. 6 Five different examples for changes in charge states of hypothetical Cu-exchanged SSZ-13 samples upon autoreduction. In pie charts, Cu(II) is shown in red, Cu(I) in blue, and  $[\text{Cu}_2\text{OH}]^{2+}$  in green.



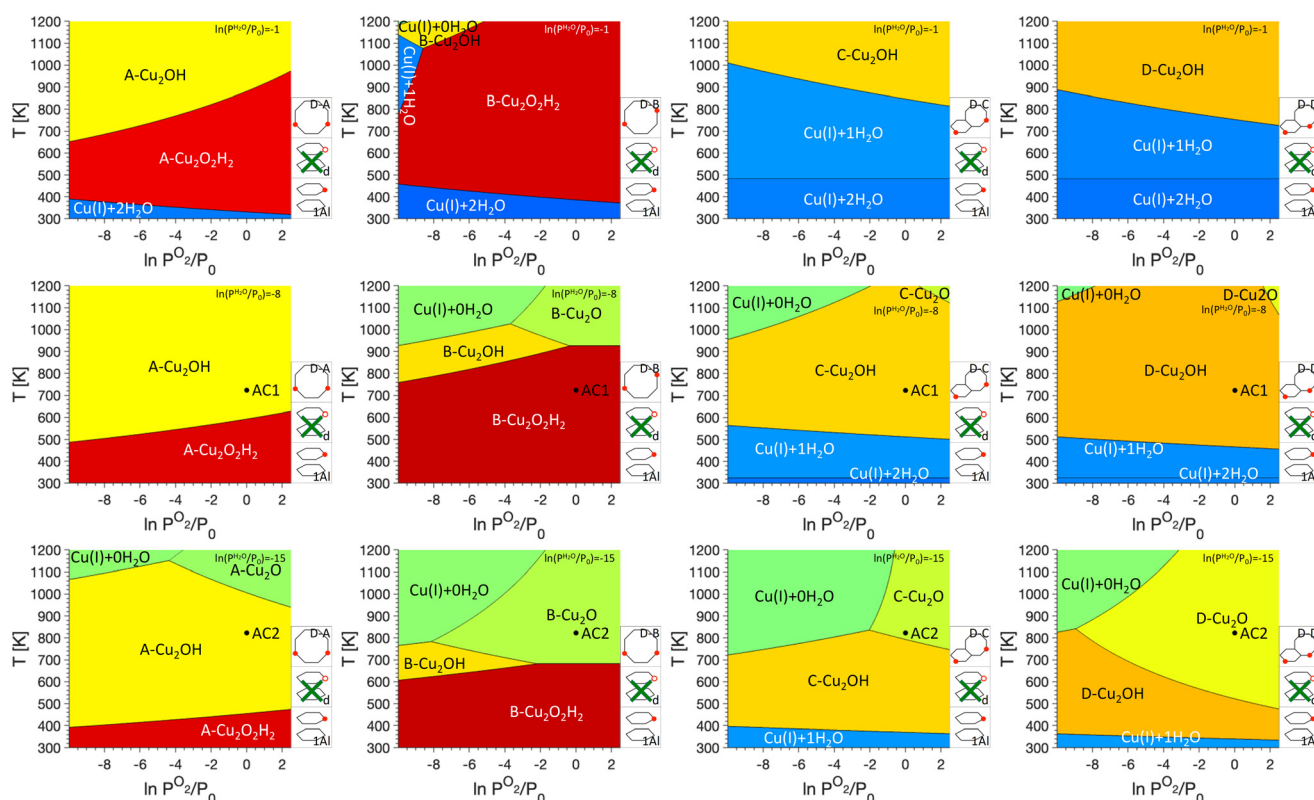


5% of all Cu, and the remaining 40% of Cu are exchanged in 1Al. For this zeolite, initially 90% of all Cu are in a +2 charge state during activation, while 5% of all Cu are present as  $[\text{Cu}_2\text{OH}]^{2+}$  and Cu(I). Upon pressure reduction, we find a total of 45% as Cu(I) and Cu(II), with 10% being present as  $[\text{Cu}_2\text{OH}]^{2+}$ . In example 4, we focus on a zeolite with a Si/Al ratio of 15 but synthesized without Al pairs.<sup>60</sup> Here, the exchange site distribution is comparable to example 3, but the 2Al-A sites are absent and 2Al-E and 1Al are increased by 10% compared to example 3. Accordingly, 10% more of the Cu can be reduced upon pressure reduction. In example 5 we focus on a zeolite with a Si/Al ratio of 8. Here, most of the Al is present in paired exchange positions, and we choose Cu in 2Al-A = 25% and Cu in 2Al-E = 25%. Additionally, all dimer exchange sites are present with 10%, which leaves 10% for Cu in the 1Al exchange position. For this material we find a Cu distribution of 80% Cu(II), 10% Cu(I), and 10%  $[\text{Cu}_2\text{OH}]^{2+}$  after activation. When the pressure is lowered, only 10% of all sites are reduced from Cu(II) to Cu(I) and 10% of all sites are reduced from Cu(II) to  $[\text{Cu}_2\text{OH}]^{2+}$ .

These examples illustrate that our model predicts a significant impact of materials parameters on autoreduction. First, we find that lowering the Si/Al ratio reduces the amount of Cu sites that can be reduced during autoreduction. We also find that synthesis parameters can impact autoreduction in two

ways, namely (i) silanol defects can be created, which are detrimental to autoreduction, and (ii) Al atoms can pair up in six rings, which again lowers the amount of Cu that can be reduced by a pressure change. In the previous paragraph we only studied examples for full ion exchange, where all exchange sites were occupied by Cu. When ion exchange levels are lowered, the least stable exchange positions will not be filled. In our case this is the 1Al position, which mainly contributes to autoreduction. Our model therefore predicts that fewer Cu atoms will be autoreduced for zeolites with lower ion exchange levels. The predictions of our model show excellent agreement with experimental measurements, which have shown that a smaller fraction of Cu atoms is autoreduced for zeolites with lower Si/Al ratios.<sup>25</sup> However, our prediction of the impact of Cu exchange levels on autoreduction does not match experimental observations for the zeolite Mordenite. However, the Cu site distribution is expected to vary significantly with zeolite framework, which in turn will significantly impact changes upon exposure to autoreduction conditions. Additionally, the impact of synthesis protocol (*i.e.*, the presence of defects) on autoreduction has been little studied so far and still requires experimental confirmation.

The work performed here also allows us to address one aspect that has been extensively discussed in the literature, namely the nature of the Cu dimers present in the zeolite



**Fig. 7**  $P^{\text{O}_2}/T$  phase diagrams for monomers anchored in 1Al and dimer/trimer anchoring configurations D-A, D-B, D-C, and D-D, without defects at different  $P^{\text{H}_2\text{O}}$  values. Regions of uniform color correspond to the site indicated by the label. Schematics of the included anchoring configurations are shown in the bottom right and  $P^{\text{H}_2\text{O}}$  values are given in the legend on the top right. Activation conditions for typical activation (AC1) and an activation protocol suggested by Rhoda *et al.*<sup>21</sup> (AC2) are marked by black circles.





SSZ-13 after activation in O<sub>2</sub> for partial methane oxidation. In previous work, phase diagrams showed that for activation at 773 K hydroxylated dimers are present<sup>36</sup> which were later characterized *via* UV-vis spectroscopy.<sup>37</sup> At the same time, after activation in O<sub>2</sub> at 873 K Rhoda *et al.* identified a Cu<sub>2</sub>O site which is active for methane activation.<sup>21</sup> Here we again use phase diagrams to rationalize these different observations and a comparison of the four dimer sites is shown in Fig. 7. Since we are interested in changes for sites anchored in the different dimer exchange positions, 1Al was chosen as reference for Cu monomers and defects were not considered in the phase diagrams. In the specific activation procedure used by Rhoda *et al.*<sup>21</sup> the material is first exposed to a He atmosphere at 973 K. This approach should allow for a thorough dehydration of the material, and we assume that the material will show a  $\ln(P^{\text{H}_2\text{O}}/P_0)$  of  $-15$ . Subsequently, the system is cooled to 823 K and exposed to O<sub>2</sub>, which we assume to be at  $\ln(P^{\text{O}_2}/P_0)$  of 0. As can be seen in Fig. 7, for D-B, D-C, and D-D, Cu<sub>2</sub>O sites are preferred and only for D-A, Cu<sub>2</sub>OH is found at the given conditions. In a conventional activation approach, the material is heated to temperatures between 723 K and 773 K and exposed to O<sub>2</sub> for extended periods of time. A dehydration step is missing, and we therefore assume that this will lead to a  $\ln(P^{\text{H}_2\text{O}}/P_0)$  of  $-8$ . Under these conditions, Cu<sub>2</sub>OH is preferred for D-A, D-C, and D-D, while Cu<sub>2</sub>O<sub>2</sub>H<sub>2</sub> is preferred for D-B.

## Conclusions

In this contribution, we used phase diagrams to investigate the changes in charge state of Cu atoms in SSZ-13 under different conditions. We investigated various Al configurations for Cu anchoring and found that specifically Cu(II)-OH sites associated with well-separated Al atoms undergo autoreduction upon a reduction in pressure at activation conditions. At the same time, specific hydroxylated dimers can undergo a reduction. Using these insights, we were then able to reproduce trends for autoreduction previously observed in the literature. Overall, this work demonstrates that Cu sites in zeolites are highly sensitive to the applied conditions and the optimization of activation and reaction conditions is necessary to arrive at active sites with optimal outcomes for a specific application.

## Conflicts of interest

There are no conflicts to declare.

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