


Cite this: *RSC Adv.*, 2024, 14, 17041

Recent advances and mechanism of plasmonic metal–semiconductor photocatalysis

Ting Kong,^a Aizhen Liao,^a Yonggang Xu,^a Xiaoshuang Qiao,^a Hanlu Zhang,^a Linji Zhang^a and Chengyun Zhang^b

Benefiting from the unique surface plasmon properties, plasmonic metal nanoparticles can convert light energy into chemical energy, which is considered as a potential technique for enhancing plasmon-induced semiconductor photocatalytic reactions. Due to the shortcomings of large bandgap and high carrier recombination rate of semiconductors, their applications are limited in the field of sustainable and clean energy sources. Different forms of plasmonic nanoparticles have been reported to improve the photocatalytic reactions of adjacent semiconductors, such as water splitting, carbon dioxide reduction, and organic pollutant degradation. Although there are various reports on plasmonic metal–semiconductor photocatalysis, the related mechanism and frontier progress still need to be further explored. This review provides a brief explanation of the four main mechanisms of plasmonic metal–semiconductor photocatalysis, namely, (i) enhanced local electromagnetic field, (ii) light scattering, (iii) plasmon-induced hot carrier injection and (iv) plasmon-induced resonance energy transfer; some related typical frontier applications are also discussed. The study on the mechanism of plasmonic semiconductor complexes will be favourable to develop a new high-performance semiconductor photocatalysis technology.

Received 16th April 2024
Accepted 10th May 2024

DOI: 10.1039/d4ra02808b

rsc.li/rsc-advances

1. Introduction

With the increasing shortage of global environmental resources, semiconductor photocatalysis has become a very promising technology and has attracted considerable attention of researchers.^{1–3} Owing to the outstanding physical and chemical performances, semiconductor catalysts can realize the effective conversion of solar energy into chemical energy, and have been widely used in environmental governance, water purification, carbon dioxide reduction and other industrial applications.^{4–6} When the light energy is larger than the band gap of the semiconductor, electron–hole (e^-/h^+) pairs will be generated on a femtosecond time scale, which will participate in redox reactions.⁷ However, the application of semiconductor materials has been greatly limited by defects of the high recombination rate of e^-/h^+ pairs, large band gap and low utilization of visible light.⁸ Recent progress has focused on improving the light absorption efficiency of semiconductor materials, inhibiting the recombination of photogenerated carriers, and enhancing the activity of catalysts.^{9–11} Therefore, designing a novel and efficient semiconductor photocatalyst is important for improving the energy conversion efficiency.

Different strategies such as vacancies, doping, surface reconstruction, heterogeneous construction, and noble metal modification have been explored to regulate the performance of semiconductor photocatalysts.^{12–14} Among them, the use of noble metal nanoparticles (NPs) (such as Au, Ag, and Cu) is a promising approach to drive a chemical reaction due to their unique feature of localized surface plasmon resonance (LSPR), and the plasmons are collective oscillations of free electrons on a metal surface driven by the incident light at or near the oscillation frequency.^{15–18} During the decay of LSPR, the energies stored in the local electromagnetic field can be reradiated *via* scattering and the oscillations can dephase into hot electron generation, resulting in the absorption of light by plasmonic NPs.^{19,20} Owing to the great light harvesting and light scattering ability, plasmonic metals provide an efficient way to enhance the photoconversion performance by transferring solar energy to adjacent semiconductors.^{21,22} For plasmon-semiconductor composite materials, plasmonic metal NPs can inhibit charge carrier recombination by plasmon-induced hot electron injection or plasmon-induced resonance energy transfer.²³ By transferring plasmonic energy or hot electrons in a direct/indirect way to adjacent semiconductors, the utilization of visible light and photocatalytic performance of semiconductors are improved.²⁴

Numerous plasmonic metal–semiconductor composites have been devoted to control plasmon-enhanced photocatalysis, while few studies have focused on generalizing the

^aSchool of Science, Xi'an University of Posts & Telecommunications, Xi'an 710121, China. E-mail: kongting0302@xupt.edu.cn; azliao@xijyou.edu.cn; ygxu@xupt.edu.cn; qiaoxs@xupt.edu.cn; zhanghl@xupt.edu.cn; zhanglinji@xupt.edu.cn

^bSchool of Electronic Engineering, Xi'an University of Posts & Telecommunications, Xi'an 710121, China. E-mail: cyzhang@xupt.edu.cn



mechanism. In this perspective, recent progress in plasmonic metal–semiconductor photocatalysis is simply discussed with emphasis on the main four mechanisms: (i) enhanced local electromagnetic field, (ii) light scattering, (iii) plasmon-induced hot carrier injection and (iv) plasmon-induced resonance energy transfer. In addition, we summarize the development of plasmonic semiconductors for photocatalysis and their prospects, hoping to provide effective theoretical guidance and approaches for the design and application of new plasmonic metal–semiconductor photocatalysts.

2. Plasmonic semiconductor photocatalysts

Since the concept of ‘plasmonic photocatalysts’ was first proposed by Awazu *et al.* in 2008, various plasmonic metal/semiconductor composites were employed to achieve plasmon-enhanced semiconductor photocatalysis, which have been widely applied in the fields of energy conversion and environmental governance.²⁵ Due to their unique LSPR property, the noble metal nanostructures can increase the light absorption ability and reduce the electron–hole recombination rate by trapping photogenerated electrons transferred from the semiconductor photocatalyst, thus promoting the reaction efficiency of photocatalysis. Au and Ag nanostructures are often used as supports for plasmon-enhanced semiconductor photocatalysis due to their excellent electromagnetic properties and visible light response. Razzari *et al.* achieved enhanced photocatalytic activity of Au–TiO₂ nanohybrids for water splitting by controlling the absorption and local electric field based on whispering gallery mode resonances.²⁶ Zhou *et al.* designed Ag@TiO₂ nanocatalysts used for single-molecule imaging, in which the catalytic activities and hot electrons generated by Ag core are affected by the TiO₂ shell thickness.²⁷ Moreover, other plasmonic metal NPs such as Pt, Cu, Al, Bi, and Co are used to achieve high photocatalytic efficiency, and have been reported in many works. For instance, Jia *et al.* prepared Pt/γ-Al₂O₃ plasmonic catalysts to effectively degrade volatile organic compounds under sunlight irradiation based on the plasmonic photothermal effect of Pt NPs.²⁸ Zeng *et al.* realized broad-spectrum absorption under near-infrared irradiation by using Cu NPs hybridized with carbon quantum dots, which possess a high catalytic efficiency for H₂ evolution due to the SPR effect of Cu NPs.²⁹ In addition, bimetallic nanostructures such as Au/Ag, Pt/Cu, and Au/Cu are developed as potential candidates for plasmon-induced semiconductor photocatalysis, which delivers enhanced catalytic performance not possible with their individual components under certain conditions.^{30–32}

The typical semiconductor material of TiO₂ as a green technology was first reported by Fujishima *et al.* in 1972 and has been widely used in the photocatalysis field.³³ However, the bandgap of TiO₂ is 3.2 eV, which only absorbs ultraviolet light and limits its applications. Then, various semiconductor materials have been designed to facilitate the application of photocatalysis in the field of environmental treatment and energy conversion, including semiconductor oxide

photocatalysts such as ZnO, Cu₂O, WO₃, and Fe₂O₃, perovskite structured ABO₃ materials such as BiTiO₃, SrTiO₃, NaTaO₃, and PrFeO₃, and novel bismuth photocatalysts such as BiOX (X = F, Cl, Br, and I), Bi₂O₃, Bi₂MO₆ (M = Cr, Mo and W), and BiVO₄.^{34–36} In addition, since the emergence of monolayer graphenes, two-dimensional (2D) nanomaterials and layered nanomaterials have been designed to promote the development of photocatalysis due to their thickness-dependent physical and chemical properties.³⁷ Although plasmonic semiconductor photocatalysts have made remarkable achievements in energy conversion, their structural design optimization, mechanism explanation and photocatalytic efficiency still need to be improved.

3. Mechanism of plasmonic metal–semiconductor photocatalysis

With the development of plasmonic photocatalysts, LSPR-enhanced photocatalysis has become a green technology to solve the problems of energy shortage and environmental pollution by using solar energy, which has attracted extensive attention in recent decades. LSPR decay can achieve significant electromagnetic field enhancement in the nanoscale “hot spot” area around the plasmonic metal NP surface, which is called near-field enhancement, leading to strong light absorption and enhanced carrier generation. For plasmonic metal semiconductor composites, four possible mechanisms of plasmon-enhanced semiconductor photocatalysis are proposed and briefly discussed below (Fig. 1).

3.1. Enhanced local electromagnetic field

For metal–semiconductor composite nanostructures, the localized field enhancement effect caused by surface plasmon decay of metal nanostructures can improve the light absorption of adjacent semiconductor materials. Due to the strong localized effect of field enhancement, the spatial distribution of photogenerated electron–hole pairs inside the semiconductor material can be regulated, and the distance of charge diffusion to the surface is shortened, thereby suppressing the carrier recombination and improving the charge separation efficiency.

Liu *et al.* reported that the photocatalytic conversion rate of hydrogen evolution using Au nanochains was significantly higher than that of conventional isolated Au NPs due to the

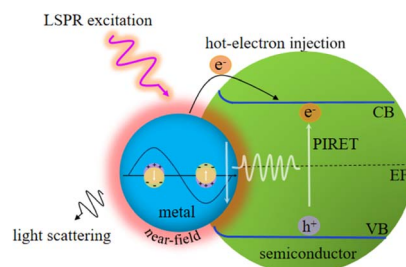


Fig. 1 Mechanism diagram for plasmonic metal–semiconductor photocatalysis.



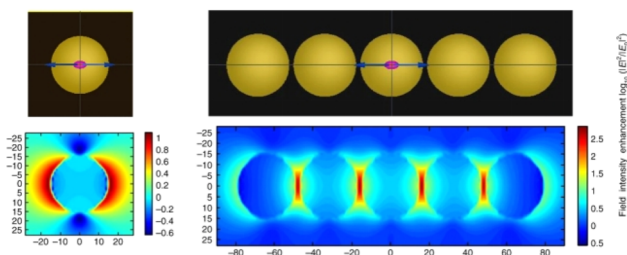


Fig. 2 Simulated near-field distributions of isolated Au NPs and Au nanochips with five coupled Au NPs using FDTD.³⁸ Reproduced from ref. 38 with permission from Springer Nature, copyright [2019].

enhanced local electromagnetic field effect (Fig. 2).³⁸ The short distance between $\text{Zn}_{0.67}\text{Cd}_{0.33}\text{S}$ semiconductors and plasmonic Au-nanochain is favourable for the transfer of plasmonic energy from metal to semiconductor. The strongly coupled Au-chain exhibits an intense local electromagnetic field, which enhances the charge carrier's formation rate and lifetime, thus improving the photocatalytic activity of semiconductors. Zou *et al.* achieved plasmon-enhanced photocatalytic reduction of CO_2 using Au@TiO_2 yolk-shell hollow spheres based on local field enhancement effect. Au nanostructures not only enhance the generation and separation efficiency of charge carriers in TiO_2 shells, but also promote the generation of high-grade carbon species, thus contributing to the improvement of photocatalytic performance.³⁹

3.2. Light scattering

Upon irradiation, the decay of surface plasmon resonance radiation induces the generation of photons, which are scattered into the adjacent semiconductor material with extremely large light absorption and light scattering cross-sections. The light trapping effect caused by plasmonic light scattering can increase the optical path in semiconductor materials, thereby enhancing the light-harvesting ability and generation rate of charge carriers.⁴⁰ Furthermore, plasmonic light absorption makes the metal nanostructure itself an efficient light-trapping centre. If the light energy captured by the plasmon can be effectively utilized, the energy conversion efficiency can be greatly improved.

Liu *et al.* designed a novel $\text{Au/Cu}_2\text{O/Pt}$ photoelectrode to achieve highly efficient photoelectrochemical water splitting due to enhanced light harvesting and photoinduced charge separation.⁴² Under light irradiation, the plasmonic hot electrons generated by the Au nanolayer were transferred to the CB of Cu_2O and increased the light-harvesting ability. A Schottky barrier was established between Cu_2O and the Pt layer, which lead to the electron transfer from Cu_2O to Pt. In addition, the rearrangement of the energy band of Cu_2O and Pt resulted in a reduction in the potential barrier and promoted the separation of electron-hole pairs, thereby enhancing the photocatalytic performance. The results of this work show that the enhanced light-harvesting ability is a synergistic effect of the SPR effect of Au and the photoinduced charge transfer effect of Pt, which provides a promising strategy for developing high-

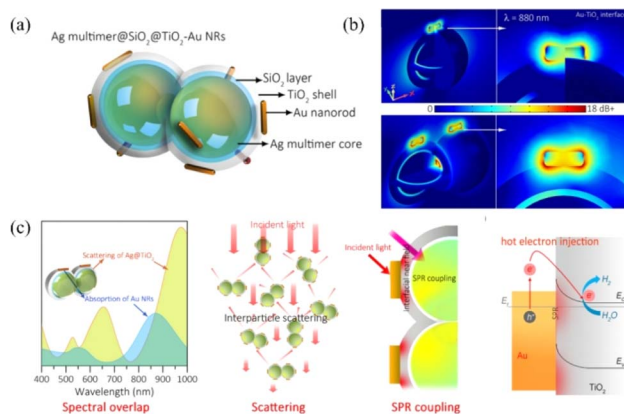


Fig. 3 Enhancement of scattering and near-field distribution of TiO_2 -Au nano hybrids using a silver resonator for plasmonic photocatalysis: (a) schematic of the Ag multimer@ SiO_2 @ TiO_2 -Au nanostructure; (b) simulated electric field enhancement distribution of Ag mono@ TiO_2 -Au and Ag multi@ TiO_2 -Au nano hybrids; (c) schematic of plasmonic Ag NP-enhanced light harvesting, including the spectral overlap, scattering, SPR coupling and hot electron injection.⁴¹ Reproduced from ref. 41 with permission from the American Chemical Society, copyright [2021].

performance plasmonic and semiconductor cocatalysts. Wang *et al.* realized highly efficient photocatalytic reduction of nitrate into N_2 by constructing a novel core-shell Ag/SiO_2 @ cTiO_2 composite.⁴³ Due to the strong light-scattering effect at the core-shell interface and the SPR effect of Ag NPs, the light-harvesting efficiency of the semiconductor is enhanced. Meanwhile, the separation efficiency of photogenerated electron-hole pairs is also enhanced by charge transfer from the TiO_2 shell to Ag NPs, which promotes the enhancement of photocatalytic performance and the stability of Ag/SiO_2 @ cTiO_2 composites. Zhang *et al.* designed a novel Ag@SiO_2 @ TiO_2 -Au plasmonic semiconductor photocatalyst system, which can achieve a higher photocatalytic activity for H_2 production due to the strong plasmonic light scattering and field enhancement effect (Fig. 3).⁴¹ Ag NPs served as a plasmonic resonator not only to increase the optical path of photocatalytic reaction *via* SPR scattering, but also to strengthen the local electric field of the TiO_2 -Au interface by Ag-Au coupling, leading to enhanced light absorption ability and separation efficiency of charge carriers, thereby boosting the catalytic activity of Ag@SiO_2 @ TiO_2 -Au nano hybrids.

3.3. Plasmon-induced hot carrier injection

The hot carriers (electrons or holes) are generated *via* radiative photon emission or nonradiative relaxation during the LSPR decay process, which play a dominant role in driving photochemical transformation.^{44,45} For the photocatalytic mechanism of the plasmonic/semiconductor composite structure, there are two possible pathways for the LSPR-induced hot electron effect: one is direct hot electron transfer, and the other is indirect hot electron transfer (tunnelling effect) (Fig. 4).¹⁰ The hot electrons with higher energy under illumination are injected into the conduction band (CB) of the adjacent semiconductor, and they



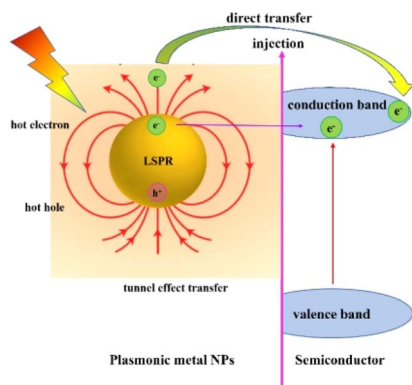


Fig. 4 Schematic of the hot electron injection mechanism for plasmonic semiconductors.

can activate specific chemical bonds in reactants and alter reaction pathways to regulate product selectivity, thereby enhancing the photocatalytic efficiency.⁴⁶

In general, the indirect hot electron transfer process is a multi-step process with a low efficiency compared to the direct hot electron transfer process, which is a one-step transfer process in metal/semiconductor composite systems. The contact between the metal and the semiconductor usually forms the Schottky barrier. Metals and other plasmons excite intra- or inter-band transitions *via* electron–electron interaction to form hot electrons. When the energy of hot electrons is greater than the Schottky barrier, they can be injected into the semiconductor across the barrier. In addition to crossing the Schottky barrier, hot electrons can be injected into the semiconductor by the tunnelling transfer effect.

Chen *et al.* investigated the shell thickness dependence of the plasmon-induced hot electron injection process by designing an Au@CdS core–shell nanocrystal for photocatalysis (Fig. 5).⁴⁷ The results indicated that the hot electron injection efficiency first increases and then decreases after a critical point around 8–10 nm with the increase in CdS shell thickness, which was mainly determined by hot-electron energy distribution of the Au core and the Schottky potential between the Au core and the CdS shell. This work provides a new approach for optimizing the photocatalytic performance of plasmonic semiconductor systems. Choudhury *et al.* discussed the photocatalytic performance of binary hybrids of Ag–TiO₂ and Au–TiO₂ and ternary plasmonic semiconductor heterostructures of Ag–TiO₂–CN and Au–TiO₂–CN in the degradation of RhB and phenol, and clarified the mechanism for enhanced photocatalysis.⁴⁸ For binary hybrids, the enhanced photocatalytic activity was attributed to the hot electron injection (direct or indirect) from Ag (or Au) to TiO₂ and localized electromagnetic field enhancement effect near the metal–semiconductor interface. As regards the ternary hybrids, it is evidenced that the enhanced photocatalytic activity is caused by hot electron injection either *via* quantum tunnelling or intermediate states with assistance of interfacial charge transfer from TiO₂ to C₃N₄, which can reduce the recombination rate of carriers and improve the ability of photodegradation of

pollutants. In addition, the Ti³⁺ states in the TiO₂ surface can increase the possibility of quantum tunnelling effect of the plasmonic electrons and facilitate direct transfer of hot electrons from the metal to the semiconductor, thereby contributing to enhanced photocatalytic efficiency.

Besides the plasmon-induced hot carrier effect, the coupling of phonons and electrons exists at the interface between the metal and the semiconductor hybrid system, which can be affected by controlling the temperature. Jacinto *et al.* studied the effect of phonons on plasmon-induced hot electron generation and charge transfer for Au/TiO₂ and Au/ZrO₂ hybrid systems under different operating temperatures.⁴⁹ The results indicated an increase in the number of hot carriers transferred from the plasmonic Au material to the TiO₂ semiconductor conduction band with the increase in temperature by transient ultrafast infrared spectroscopy technology. The increase in hot carrier injection is attributed to the phonon-enhanced hot carrier generation (phonons transfer momentum to electrons) or phonon-assisted hot carrier transfer across the Au/TiO₂ interface barrier (phonons transfer energy to hot carriers). As the height of the Schottky barrier increases with the increase in temperature, the electron injection efficiency is reduced. Therefore, due to phonon-assisted hot electron generation, the plasmon-induced hot carrier injection of the Au NP/TiO₂ hybrid system increases with the increase in temperature. This work provides a new approach for the application of plasmonic devices.

3.4. Plasmon-induced resonance energy transfer

Plasmon-induced resonance energy transfer (PIRET) refers to the transfer of trapped energy by surface plasmons *via* dipole–dipole interactions to adjacent semiconductors.^{50,51} PIRET is a non-radiative relaxation channel of surface plasmons, which can be observed on strong coupling between plasmons and excitons.⁵² The Förster resonance energy transfer (FRET) effect is a similar process commonly used to describe the energy transfer between two chromophores (fluorescent dye molecules or semiconductor quantum dots) in which the donor excited state energy can be transferred to the receptor.⁵³ However, FRET is a weakly coupled and incoherent process that occurs after the thermal relaxation of the excited state. The PIRET in the plasmon-semiconductor hybrid system is an ultrafast coherent process, which can achieve a higher energy conversion efficiency in photocatalytic applications.⁵⁴

Wu *et al.* elaborated the mechanism of PIRET by employing the Au@SiO₂@Cu₂O core–shell NPs and distinguished the theory of PIRET from FRET.⁵⁴ The energy transfer between plasmon and semiconductor depends on the dipole moments. Under dipole–dipole coupling, if the plasmon is excited, which the energy is transferred to the Cu₂O semiconductor by PIRET, conversely, the energy is transferred from the Cu₂O semiconductor to the plasmon by FRET. In addition, the PIRET efficiency can be controlled by spatial distance and spectral overlap between the plasmonic Au nanostructure and the Cu₂O semiconductor. The experimental results indicate that a thin dielectric layer of SiO₂ can inhibit electron transfer and provide



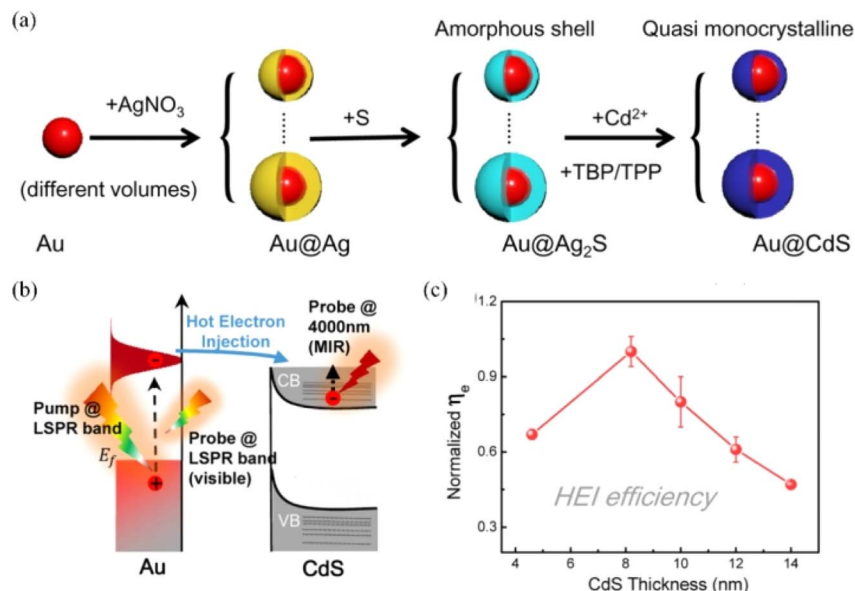


Fig. 5 Plasmon-induced hot electron injection in the Au@CdS core-shell nanocrystal: (a) schematic of the preparation process of the Au@CdS core-shell nanocrystal; (b) schematic of the hot electron injection process between Au and CdS under pump LSPR band excitation; and (c) hot electron injection efficiency as a function of CdS thickness.⁴⁷ Reproduced from ref. 47 with permission from the American Chemical Society, copyright [2021].

the largest PIRET enhancement due to the LSPR effect of Au and the PIRET effect. Moreover, PIRET is an ultrafast and coherent process that occurs before the thermal relaxation of the plasmon, and it can effectively promote light absorption by utilizing the energy near and below the band gap of semiconductors.

In addition, Wu *et al.* designed metal@TiO₂ core-shell NPs in order to control the PIRET and HEI process by changing the

spectral overlap and introducing SiO₂ layer (Fig. 6).²³ The relative increase in carrier density for PIRET depends mainly on the spectral overlap between plasmons and semiconductors and HEI mainly depends on plasmon absorption. The extinction spectra showed that the spectral overlap for Ag@TiO₂ was significantly enhanced compared to that of pure TiO₂, while there is no spectral overlap for plasmonic Au NPs with the interband transitions dominated below the wavelength of 500 nm (Fig. 6b). Both the PIRET and HEI processes are favorable to enhance the photoconversion for metal/TiO₂ semiconductor systems (Fig. 6c). Moreover, the transient absorption spectroscopy was performed to study plasmonic the enhancement mechanism (Fig. 6d). The results demonstrated that both the PIRET and HEI processes occurred for the Ag@TiO₂ system, while for Au@TiO₂, only HEI transferred from Au to TiO₂. Due to the spectral overlap, the PIRET can occur for Ag@SiO₂@TiO₂ NPs. There is no PIRET or HEI occurring for Au@SiO₂@TiO₂ NPs. This work contributes to better understand the mechanism of plasmon-enhanced light conversion.

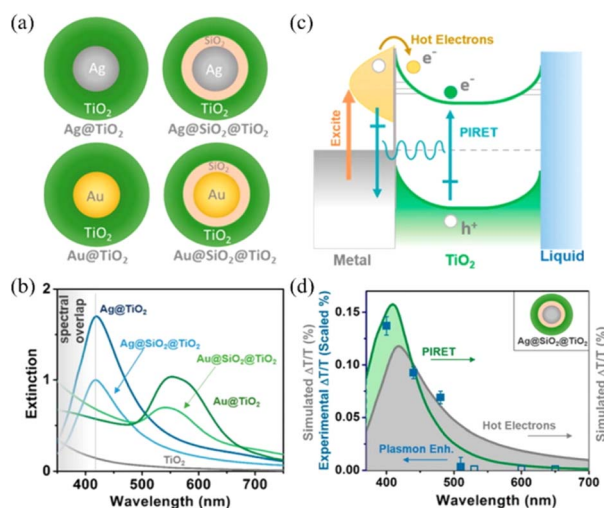


Fig. 6 Controlling PIRET and HEI for metal@TiO₂ core-shell NPs: (a) schematic of the metal@TiO₂ core-shell NPs; (b) extinction spectra of the metal@TiO₂ NPs; (c) schematic of the PIRET and HEI mechanism of the metal@TiO₂ system; and (d) experimental and simulated change of transmission for Ag@SiO₂@TiO₂ as a function of wavelength.²³ Reproduced from ref. 23 with permission from the American Chemical Society, copyright [2015].

4. Applications

Benefiting from the LSPR effect of plasmonic NPs, the absorption cross-section of semiconductors, and light absorption range and electron-hole charge separation rate of semiconductor can be effectively improved, thus expanding the applications of plasmonic semiconductors in the field of photocatalysis. In this section, we discuss the recent progress in plasmon-enhanced semiconductor photocatalysis for various applications including energy conversion and environmental remediation. The various applications of typical plasmonic metal-semiconductor photocatalysts are listed in Table 1.

Table 1 Typical plasmonic metal–semiconductor photocatalysts for a variety of photocatalytic applications

Plasmonic metal-semiconductors	Photocatalytic applications	Published year	Reference
Au–TiO ₂	Photocatalytically oxidize ethanol and methanol; photovoltaic fuel cell	2005	55
Ag/SiO ₂ –TiO ₂	Decomposition of MB	2008	56
Au@CeO ₂	Oxidation of benzyl alcohol to benzaldehyde	2014	57
Au/ZnO	MO dye degradation	2016	58
Au/Cu ₂ O/Pt	Water splitting	2018	42
Au chain@Zn _{0.67} Cd _{0.33} S	Photocatalytic hydrogen evolution	2019	38
Cu-doped ZnO–CuO	Photo-decontamination of textile effluents	2020	59
Mg/MgO	SF ₆ environmental remediation	2020	60
Bi/α-MnS/Bi ₂ MoO ₆	CO ₂ reduction	2021	5
Ag/SiO ₂ @cTiO ₂	Photocatalytic reduction of nitrate	2021	43
Ag/BiFeO ₃ fibrous	Photoelectrochemical energy conversion	2021	61
SiO ₂ –Au–Pt	Degradation of MB	2021	62
Ag/BiVO ₄	Photodegradation of RhB	2022	63
Au NBP@Cu ₂ O	Treatment of breast cancer	2023	64
Bi/TiO ₂	CO ₂ reduction	2024	65

4.1. Energy application

The photocatalytic water-splitting process based on plasmonic metal–semiconductors enables obtaining an effective, clean, and environmentally friendly renewable energy source, which has been widely studied by many researchers in recent years with the aim of improving photocatalytic hydrogen production performance.⁶⁶ Liu *et al.* realized plasmon-enhanced photocatalytic water splitting with a Au NPs/TiO₂ composite under visible light irradiation (Fig. 7).⁶⁷ The results show that the photocurrent of TiO₂ can be enhanced by 66 times with plasmonic Au NPs at 633 nm excitation, indicating that the photocatalytic water splitting ability is significantly enhanced due to the local field enhancement effect. Yoon *et al.* provided an approach for photocatalytic water splitting by designing a Ag NP-impregnated BiVO₄ nanocomposite (Fig. 8).⁶⁸ The photoelectrochemical measurement and the simulation results show that the catalytic performance of Ag/BiVO₄ nanocomposites is significantly higher than that of original materials, which is attributed to the increased carrier generation and charge separation efficiency caused by the LSPR effect.

Plasmon-driven semiconductor photocatalytic carbon dioxide (CO₂) reduction to hydrocarbon fuels has attracted

widespread attention as a promising strategy for producing solar fuels and reducing greenhouse gases, which needs the design of highly efficient photocatalysts that can be boosted by the broad spectrum of wavelengths of sunlight.³⁹ Long *et al.* reported the broadband CO₂ reduction reaction with water to hydrocarbons by plasmonic Au rod@CuPd₂ nanostructures, which provided a new strategy for energy conversion with near-infrared photons (Fig. 9).⁶⁹ In this work, a co-catalyst of Au nanorods with CuPd₂ shells was designed to realize an enhanced plasmon-driven photocatalytic efficiency. The yield rate of CH₄ is 0.55 mmol g^{−1} h^{−1} under full-spectrum light with 100% selectivity for hydrocarbons and the apparent quantum efficiency (AQE) for Au rod@CuPd₂ is 0.38% under 800 nm light irradiation (Fig. 9c and d). The X-ray photoelectron spectroscopy and theoretical calculation results show that the local electric field and electron transfer together promote the CO₂ reduction reaction by effectively utilizing low-energy near-infrared light. Li *et al.* achieved plasmon-enhanced photocatalytic CO₂ reduction efficiency for ammoniated α-MnS/Bi₂MoO₆ S-scheme heterostructures by the SPR effect of plasmonic Bi, which provide a novel insight into the field of plasmonic metal–semiconductor photocatalysis for energy conversion.⁵

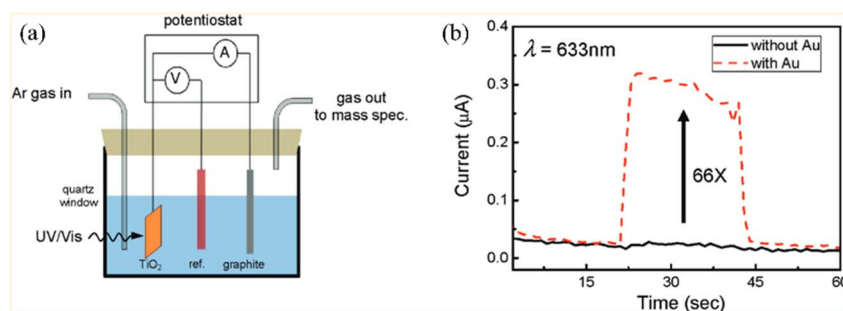


Fig. 7 Applications of plasmon-enhanced photocatalytic water splitting with Au NP/TiO₂ composites: (a) schematic of the TiO₂ photocatalytic water-splitting experimental device and (b) photocurrent of TiO₂ with and without Au NPs under 633 nm excitation.⁶⁷ Reproduced from ref. 67 with permission from the American Chemical Society, copyright [2011].



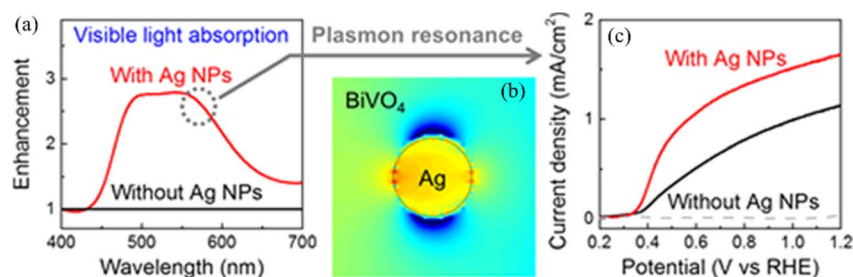


Fig. 8 Applications of plasmon-enhanced photocatalytic water splitting with a Ag NP-impregnated BiVO₄ composite: (a) absorption enhancement of BiVO₄ with and without Ag NPs; (b) simulated light absorption distribution of BiVO₄ with a Ag NP; and (c) current density as a function of the potential of BiVO₄ with and without Ag NPs.⁶⁸ Reproduced from ref. 68 with permission from the American Chemical Society, copyright [2018].

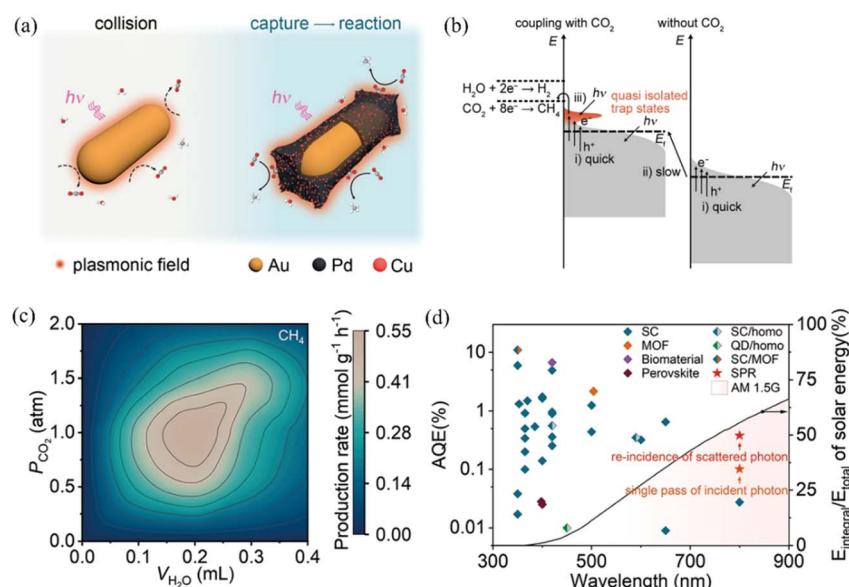


Fig. 9 Applications of plasmon-induced CO₂ reduction with Au rod@CuPd₂: (a) schematic of the plasmon-induced CO₂ reduction for pure Au rods (left) and Au@CuPd₂ (right); (b) schematic of the mechanism of plasmon-induced CO₂ reduction reaction, (c) production rate of CH₄ under different CO₂ pressures and H₂O volumes and (d) AQE of Au rod@CuPd₂.⁶⁹ Reproduced from ref. 69 with permission from Springer Nature, copyright [2023].

Ammonia as a carbon-free solar energy storage material can be manufactured from N₂ or water. Plasmon-induced photocatalytic nitrogen fixation is considered as a promising green artificial technology for green ammonia production.^{70–72} Fang *et al.* achieved high-performance nitrogen fixation driven by plasmonic Au/Ru electrocatalysts.⁷³ Due to the inertness of plasmonic AuNPs, Ru acts as a bridge to promote the effective utilization of hot electrons by transferring directly from Au to N₂, thereby enhancing the plasmon-induced N₂ fixation rate. Moreover, Zhang *et al.* utilized porous CuFe to achieve the application of plasmonic Cu photocatalysts in N₂ photofixation.⁷⁴

4.2. Environmental application

Besides, plasmonic metal–semiconductor photocatalysis has achieved great progress in other applications such as environmental governance, and cancer treatment.^{75–77} Haghighi

et al. achieved high-performance MB photodegradation by designing an ultra-porous Cu plasmon-induced CuZnO–CuO heterojunction.⁷⁸ The experiment results indicated that when the fuel ratio increases from 0.5 to 2, the degradation ability of the dye was enhanced significantly due to the increase in charge carrier separation and SPR mechanism of Cu NPs. Sun *et al.* designed a plasmonic Z-scheme Cu₂O/Bi/Bi₂MoO₆ heterojunction, which exhibits an excellent photocatalytic activity in the degradation of sulfadiazine and Ni(II) due to the SPR effect.⁷⁹ Gutiérrez *et al.* first designed a plasmonic Mg/MgO NP platform with controllable LSPR effect, which presents high-performance photocatalytic properties for environmental remediation by degrading SF₆ into non-toxic products of MgF₂ and MgSO₄.⁶⁰ Chang *et al.* demonstrated that Au NBP@Cu₂O nanozymes can effectively treat breast cancer under a near-infrared II window due to the plasmon-enhanced catalytic activity.⁶⁴

5. Conclusions

Plasmon-enhanced photocatalysis has been a hot topic in recent years due to its unique surface plasmon properties. In this review, the main mechanism of plasmonic metal-semiconductor photocatalysis and its recent progress were briefly discussed. As for the plasmonic semiconductor composites, the localized field enhancement effect caused by the LSPR decay of metal nanostructures can improve the light absorption of adjacent semiconductor materials. The strong plasmonic light-harvesting and light-scattering ability makes the metal nanostructure itself become an efficient light-trapping centre, which plays a significant role in broadening the absorption wavelength range and improving the photocatalytic activity of semiconductor photocatalysts. The light energy stored in plasmonic metal NPs can be transferred to adjacent semiconductor materials that can maximize solar energy conversion. Moreover, plasmonic metal nanostructures can improve the photocatalytic performance of semiconductors by enlarging the light absorption region or enhancing the separation efficiency of electron-hole pairs by plasmon-induced hot electron injection or plasmon-induced resonance energy transfer.

Although great progress has been made in plasmonic metal-semiconductor photocatalysis, the mechanism and other related issues need further systematic and in-depth study. For example, (i) designing new plasmonic nanomaterials: in traditional plasmonic metal-semiconductor photocatalysis systems, the plasmonic materials are usually noble metals such as Au and Ag, but they have limitations such as high cost, instability, and low utilization of visible light. Therefore, it is crucial to find new plasmonic materials that are economic and stable with low loss to achieve enhanced photocatalytic performance. (ii) Charge/energy transfer mechanism at the metal/semiconductor interface: it is well known that the mechanisms of plasmon-induced hot electron transfer and energy transfer are of great significance to explain the mechanism of plasmon-enhanced semiconductor photocatalysis. However, thus far, this interpretation is not very clear and is based on indirect proofs. Therefore, new ideas and technologies for directly characterizing the separation of electron-hole pairs and energy transfer between plasmons and semiconductors are still facing great challenges. (iii) Explore other synergy mechanism for plasmonic metal-semiconductor photocatalysis: based on the surface plasmon effect, not only the near-field enhancement, far-field scattering, charge transfer and energy transfer effects, but also the interaction of electrons and phonons and the photothermal effect will enhance the plasmonic metal-semiconductor photocatalysis. The synergy between them, as well as new mechanisms and insights, needs further experimental and theoretical exploration and discussion. (iv) Extensive research is required for plasmonic metal-semiconductor systems in order to expand the industrial application and commercial value with low cost, high stability and catalytic efficiency. It is believed that solving the above-mentioned problems will provide a new way for the development and application of plasmonic metal-semiconductor photocatalysis.

Author contributions

T. K. wrote and prepared the manuscript for submission; A. L. and Y. X. were involved in literature search and study design. X. Q., H. Z. and L. Z. contributed to the data analysis and discussion of the contents. C. Z. reviewed and supervised the manuscript. All authors have read and agreed to the published version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (Grant 12204376, 12104366 and 22005243); Shaanxi Provincial Natural Science Foundation (Grant 2021JQ-700 and 2021JQ-699); Scientific Research Program Funded by Shaanxi Provincial Education Department (Program No. 20JK0905).

References

- 1 M. M. Abouelela, G. Kawamura and A. Matsuda, *J. Cleaner Prod.*, 2021, **294**, 126200.
- 2 B. Liu, H. Wu and I. P. Parkin, *ACS Omega*, 2020, **5**, 14847–14856.
- 3 X. Sun, H. Huang, Q. Zhao, T. Ma and L. Wang, *Adv. Funct. Mater.*, 2020, **30**, 1910005.
- 4 H. Wang, X. Li, X. Zhao, C. Li, X. Song, P. Zhang, P. Huo and X. Li, *Chin. J. Catal.*, 2022, **43**, 178–214.
- 5 L. Liu, K. Dai, J. Zhang and L. Li, *J. Colloid Interface Sci.*, 2021, **604**, 844–855.
- 6 M. Khademalrasool, M. Farbod and M. D. Talebzadeh, *Adv. Powder Technol.*, 2021, **32**, 1844–1857.
- 7 J. Schneider, M. Matsuoka, M. Takeuchi, J. Zhang, Y. Horiuchi, M. Anpo and D. W. Bahnemann, *Chem. Rev.*, 2014, **114**, 9919–9986.
- 8 T. Kong, X. Wei, G. Zhu and Y. Huang, *J. Mater. Sci.*, 2017, **53**, 708–715.
- 9 S. Li, P. Miao, Y. Zhang, J. Wu, B. Zhang, Y. Du, X. Han, J. Sun and P. Xu, *Adv. Mater.*, 2021, **33**, e2000086.
- 10 P. Jiménez-Calvo, V. Caps and V. Keller, *Renewable Sustainable Energy Rev.*, 2021, **149**, 111095.
- 11 D. Liu and C. Xue, *Adv. Mater.*, 2021, **33**, 2005738.
- 12 T. Kong, X. Wei, G. Zhu and Y. Huang, *Chin. J. Phys.*, 2017, **55**, 331–341.
- 13 T. Kong, X. Wei, G. Zhu and Y. Huang, *J. Mater. Sci.*, 2017, **52**, 5686–5695.
- 14 Y. Li, M. Wen, Y. Wang, G. Tian, C. Wang and J. Zhao, *Angew. Chem., Int. Ed.*, 2021, **60**, 910–916.
- 15 T. Kong, C. Zhang, X. Gan, F. Xiao, J. Li, Z. Fu, Z. Zhang and H. Zheng, *J. Mater. Chem. C*, 2020, **8**, 4338–4342.
- 16 S. Linic, U. Aslam, C. Boerigter and M. Morabito, *Nat. Mater.*, 2015, **14**, 567–576.



- 17 T. Kong, C. Zhang, J. Lu, B. Kang, Z. Fu, J. Li, L. Yan, Z. Zhang, H. Zheng and H. Xu, *Nanoscale*, 2021, **13**, 4585–4591.
- 18 J. Dong, Z. Zhang, H. Zheng and M. Sun, *Nanophotonics*, 2015, **4**, 472–490.
- 19 C. Zhang, J. Qi, Y. Li, Q. Han, W. Gao, Y. Wang and J. Dong, *Nanomaterials*, 2022, **12**, 1329.
- 20 J. Dong, Y. Cao, Q. Han, Y. Wang, M. Qi, W. Zhang, L. Qiao, J. Qi and W. Gao, *Opt. Express*, 2020, **28**, 20817–20829.
- 21 M. Wang, M. Ye, J. Iocozzia, C. Lin and Z. Lin, *Adv. Sci.*, 2016, **3**, 1600024.
- 22 U. Aslam, V. G. Rao, S. Chavez and S. Linic, *Nat. Catal.*, 2018, **1**, 656–665.
- 23 S. K. Cushing, J. T. Li, J. Bright, B. T. Yost, P. Zheng, A. D. Bristow and N. Q. Wu, *J. Phys. Chem. C*, 2015, **119**, 16239–16244.
- 24 M. Fang, X. Tan, Z. Liu, B. Hu and X. Wang, *Research*, 2021, **2021**, 9794329.
- 25 K. Awazu, M. Fujimaki, C. Rockstuhl, J. Tominaga, H. Murakami, Y. Ohki, N. Yoshida and T. Watanabe, *J. Am. Chem. Soc.*, 2008, **130**, 1676–1680.
- 26 J. Zhang, X. Jin, P. I. Morales-Guzman, X. Yu, H. Liu, H. Zhang, L. Razzari and J. P. Claverie, *ACS Nano*, 2016, **10**, 4496–4503.
- 27 Y. Liu, K. Zhang, X. Tian, L. Zhou, J. Liu and B. Liu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 7680–7687.
- 28 S.-C. Cai, J.-J. Li, E.-Q. Yu, X. Chen, J. Chen and H.-P. Jia, *ACS Appl. Nano Mater.*, 2018, **1**, 6368–6377.
- 29 P. Zhang, T. Song, T. Wang and H. Zeng, *Appl. Catal., B*, 2017, **206**, 328–335.
- 30 S. I. Naya, M. Fujishima and H. Tada, *Catalysts*, 2019, **9**, 745.
- 31 Z. Bielan, A. Sulowska, S. Dudziak, K. Siuzdak and Z. Anna, *Catalysts*, 2020, **10**, 672.
- 32 J. C. Durán-Ivarez, E. Avella, R. Ramírez-Zamora and R. Zanella, *Catal. Today*, 2016, **266**, 175–187.
- 33 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- 34 K. Sharma, V. Dutta, S. Sharma, P. Raizada, A. Hosseini-Bandegharai, P. Thakur and P. Singh, *J. Ind. Eng. Chem.*, 2019, **78**, 1–20.
- 35 K. S. Schanze, P. V. Kamat, P. Yang and J. Bisquert, *ACS Energy Lett.*, 2020, **5**, 2602–2604.
- 36 A. B. Djurišić, Y. H. Leung and A. M. Ching Ng, *Mater. Horiz.*, 2014, **1**, 400.
- 37 B. Luo, G. Liu and L. Wang, *Nanoscale*, 2016, **8**, 6904–6920.
- 38 G. Yu, J. Qian, P. Zhang, B. Zhang, W. Zhang, W. Yan and G. Liu, *Nat. Commun.*, 2019, **10**, 4912.
- 39 W. Tu, Y. Zhou, H. Li, P. Li and Z. Zou, *Nanoscale*, 2015, **7**, 14232–14236.
- 40 R. A. Pala, J. S. Q. Liu, E. S. Barnard, D. Askarov, E. C. Garnett, S. Fan and M. L. J. N. C. Brongersma, *Nat. Commun.*, 2013, **4**, 1–7.
- 41 M. Liu, X. Jin, S. Li, J. B. Billeau, T. Peng, H. Li, L. Zhao, Z. Zhang, J. P. Claverie, L. Razzari and J. Zhang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 34714–34723.
- 42 D. Chen, Z. Liu, Z. Guo, W. Yan and Y. Xin, *J. Mater. Chem. A*, 2018, **6**, 20393–20401.
- 43 Z. Hou, J. Chu, C. Liu, J. Wang, A. Li, T. Lin and C. P. François-Xavier, *Chem. Eng. J.*, 2021, **415**, 1228863.
- 44 J. G. Liu, H. Zhang, S. Link and P. Nordlander, *ACS Photonics*, 2017, **5**, 2584–2595.
- 45 M. L. Brongersma, N. J. Halas and P. Nordlander, *Nat. Nanotechnol.*, 2015, **10**, 25–34.
- 46 Z. Lou, S. Kim, M. Fujitsuka, X. Yang, B. Li and T. Majima, *Adv. Funct. Mater.*, 2018, **28**, 1706969.
- 47 H. Dong, J. Feng, J. Liu, X. Wan, J. Zhang, Z. Wang, H. Chen and Y.-X. Weng, *J. Phys. Chem. C*, 2021, **125**, 19906–19913.
- 48 K. K. Paul, P. K. Giri, H. Sugimoto, M. Fujii and B. Choudhury, *Sol. Energy Mater. Sol. Cells*, 2019, **201**, 110053.
- 49 Y. Hattori, J. Meng, K. Zheng, A. Meier de Andrade, J. Kullgren, P. Broqvist, P. Nordlander and J. Sa, *Nano Lett.*, 2021, **21**, 1083–1089.
- 50 Y. Choi, Y. Park, T. Kang and L. P. Lee, *Nat. Nanotechnol.*, 2009, **4**, 742–746.
- 51 G. L. Liu, Y.-T. Long, Y. Choi, T. Kang and L. P. Lee, *Nat. Methods*, 2007, **4**, 1015–1017.
- 52 P. Vasa, W. Wang, R. Pomraenke, M. Lammers, M. Maiuri, C. Manzoni, G. Cerullo and C. Lienau, *Nat. Photonics*, 2013, **7**, 128–132.
- 53 R. Heim and R. Y. Tsien, *Curr. Biol.*, 1996, **6**, 178–182.
- 54 J. Li, S. K. Cushing, F. Meng, T. R. Senty, A. D. Bristow and N. Wu, *Nat. Photonics*, 2015, **9**, 601–607.
- 55 Y. Tian and T. Tatsuma, *J. Am. Chem. Soc.*, 2005, **127**, 7632–7637.
- 56 K. Awazu, M. Fujimaki, C. Rockstuhl, J. Tominaga, H. Murakami, Y. Ohki, N. Yoshida and T. Watanabe, *J. Am. Chem. Soc.*, 2008, **130**, 1676–1680.
- 57 B. Li, T. Gu, T. Ming, J. Wang, P. Wang, J. Wang and J. C. Yu, *ACS Nano*, 2014, **8**, 8152–8162.
- 58 J. Lu, H. Wang, D. Peng, T. Chen, S. Dong and Y. Chang, *Phys. E*, 2016, **78**, 41–48.
- 59 E. Abbasi, M. Haghighi, R. Shokrani and M. Shabani, *Mater. Res. Bull.*, 2020, **129**, 110880.
- 60 Y. Gutierrez, M. M. Giangregorio, F. Palumbo, F. Gonzalez, A. S. Brown, F. Moreno and M. Losurdo, *Nano Lett.*, 2020, **20**, 3352–3360.
- 61 J. Xu, T. Qin, W. Chen, J. Lv, X. Zeng, J. Sun, Y.-y. Li and J. Zhou, *Nano Energy*, 2021, **89**, 106317.
- 62 J. U. Salmón-Gamboa, M. Romero-Gómez, D. J. Roth, A. V. Krasavin, P. Wang, W. Dickson and A. V. Zayats, *Nanoscale Adv.*, 2021, **3**, 767–780.
- 63 S. Bakhtiarnia, S. Sheibani, E. Aubry, H. Sun, P. Briois and M. A. P. Yazdi, *Appl. Surf. Sci.*, 2022, **580**, 152253.
- 64 L. Zhao, Z. Q. Sun, Y. Wang, J. Huang, H. T. Wang, H. Li, F. Chang and Y. Y. Jiang, *Acta Biomater.*, 2023, **170**, 496–506.
- 65 W. He, J. Xiong, Z. Tang, Y. Wang, X. Wang, H. Xu, Z. Zhao, J. Liu and Y. Wei, *Appl. Catal., B*, 2024, **344**, 123651.
- 66 T. Jafari, E. Moharrerri, A. Amin, R. Miao, W. Song and S. Suib, *Molecules*, 2016, **21**, 900.
- 67 Z. Liu, W. Hou, P. Pavaskar, M. Aykol and S. B. Cronin, *Nano Lett.*, 2011, **11**, 1111–1116.
- 68 S. Y. Jeong, H.-M. Shin, Y.-R. Jo, Y. J. Kim, S. Kim, W.-J. Lee, G. J. Lee, J. Song, B. J. Moon, S. Seo, H. An, S. H. Lee,



- Y. M. Song, B.-J. Kim, M.-H. Yoon and S. Lee, *J. Phys. Chem. C*, 2018, **122**, 7088–7093.
- 69 C. Hu, X. Chen, J. Low, Y.-W. Yang, H. Li, D. Wu, S. Chen, J. Jin, H. Li, H. Ju, C.-H. Wang, Z. Lu, R. Long, L. Song and Y. Xiong, *Nat. Commun.*, 2023, **14**, 221.
- 70 L.-W. Chen, Y.-C. Hao, Y. Guo, Q. Zhang, J. Li, W.-Y. Gao, L. Ren, X. Su, L. Hu and N. Zhang, *J. Am. Chem. Soc.*, 2021, **143**, 5727–5736.
- 71 T.-A. Bu, Y.-C. Hao, W.-Y. Gao, X. Su, L.-W. Chen, N. Zhang and A.-X. Yin, *Nanoscale*, 2019, **11**, 10072–10079.
- 72 H. Huang, S. Wang, X. Fan, D. Philo, L. Fang, W. Tu, T. Qiu, Z. Zou and J. Ye, *ACS Sustain. Chem. Eng.*, 2023, **11**, 10993–11001.
- 73 H. Yin, J. Hu, C. Fang, Y. Wang, L. Ma, N. Zhang, S. Zhang, R. Jiang and J. Wang, *Nano Res.*, 2022, **16**, 360–370.
- 74 T. Hou, L. Chen, Y. Xin, W. Zhu, C. Zhang, W. Zhang, S. Liang and L. Wang, *ACS Energy Lett.*, 2020, **5**, 2444–2451.
- 75 T. T. Yang, W. T. Chen, Y. J. Hsu, K. H. Wei and T. W. Lin, *J. Phys. Chem. C*, 2010, **114**, 11414–11420.
- 76 N. Zhou, V. López-Puente, Q. Wang, L. Polavarapu, I. Pastoriza-Santos and Q.-H. Xu, *RSC Adv.*, 2015, **5**, 29076–29097.
- 77 Y. Li, S. Li, C. He, C. Zhu, Q. Li, X. Li, K. Liu and X. Zeng, *J. Alloys Compd.*, 2021, 867.
- 78 E. Abbasi, M. Haghighi, R. Shokrani and M. Shabani, *Mater. Res. Bull.*, 2020, **129**, 110880.
- 79 X. Xu, L. Meng, Y. Dai, M. Zhang, C. Sun, S. Yang, H. He, S. Wang and H. Li, *J. Hazard. Mater.*, 2020, **381**, 120953.

