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Vacancy- and doping-mediated electronic and magnetic properties of PtSSe monolayer towards optoelectronic and spintronic applications†

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Developing new multifunctional two-dimensional (2D) materials with two or more functions has been one of the main tasks of materials scientists. In this work, defect engineering is explored to functionalize PtSSe monolayer with feature-rich electronic and magnetic properties. Pristine monolayer is a non-magnetic semiconductor 2D material with a band gap of 1.52(2.31) eV obtained from PBE(HSE06)-based calculations. Upon creating single Pt vacancy, the half-metallic property is induced in PtSSe monolayer with a total magnetic moment of 4.00 μ_B . Herein, magnetism is originated mainly from S and Se atoms around the defect site. In contrast, single S and Se vacancies preserve the non-magnetic nature. However, the band gap suffers of considerable reduction of the order of 67.11% and 48.68%, respectively. The half-metallicity emerges also upon doping with alkali metals (Li and Na) with total magnetic moment of 1.00 μ_B , while alkaline earth impurities (Be and Mg) make new diluted magnetic semiconductor materials from PtSSe monolayer with total magnetic moment of 2.00 μ_B . In these cases, magnetic properties are produced mainly by Se atoms closest to the doping site. In addition, doping with P and As atoms at chalcogen sites is also investigated. Except for the half-metallic As_{Se} system (As doping at Se site), the diluted magnetic semiconductor behavior is obtained in the remaining cases. Spin density results indicate key role of the VA-group impurities in magnetizing PtSSe monolayer. In these cases, total magnetic moments between 0.99 and 1.00 μ_B are obtained. Further Bader charge analysis implies the charge loser role of all impurities that transfer charge to the host monolayer. Results presented in this work may suggest promises of the defected and doped Janus PtSSe structures for optoelectronic and spintronic applications.

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1. Introduction

Since the first successful isolation of graphene from graphite by Novoselov *et al.*,¹ extensive theoretical and experimental investigations have asserted its intriguing physical, chemical, and mechanical properties that are quite different from and superior than those of graphite. For instance, graphene has a high electron mobility ($\sim 15,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$), high absorption rate

about 2.3% of incident light, high mechanical stability - a hundred times stronger than steel,^{2,3} among others. The graphene band structure exhibits a direct point with linear dispersion at *K* point, where the valence band and conduction band meet. Practically, this zero-gap feature hinders considerably the efficiency of graphene when it is applied in electronic nanodevices.⁴ Although the graphene band gap can be opened by different methods as surface functionalization^{5,6} and cutting edges,^{7,8} it is very important to search for new semiconductor two-dimensional (2D) alternatives. So far, research efforts have conducted to the discovery of a large variety of 2D materials, which can be prepared in experiments by either top-down method (exfoliation from layered bulk structures)^{9,10} or bottom-up method (deposition growth from precursors).^{11,12} The discovered 2D materials hold great potential for catalysis,^{13,14} optoelectronics and electronics,^{15,16} photonics,^{17,18} energy conversion and storage,^{19,20} environmental and biomedical applications,^{21,22} spintronics,^{23,24} among others.

As one of the big and relevant 2D materials family, transition metal dichalcogenides (TMDs) have gained special research attention.²⁵ With a 1 : 2 stoichiometry and chemical formula

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MX_2 (M = transition metals; X = chalcogen atoms), TMDs structure is formed by sandwiching one M atomic sublayer between two X atomic sublayers. Initially, first structures based on VIB transition metals (M = Mo and W) were studied. These 2D structures exhibit semiconductor nature with relatively large energy gaps.²⁶ Moreover, other intriguing properties have been also characterized including good environmental stability, flexibility, and tunable electronic band gap, and effective properties modification.^{27,28} 2D MoX_2 and WX_2 structures have been explored as potential materials for flexible photonics,^{29,30} thermoelectrics,^{31,32} electronics,^{33,34} and light emissions,^{35,36} and so on. From the sandwiched structure, a new family called “Janus” structure MXY (X \neq Y) have emerged with the discovery of MoSSe monolayer by either substituting S sublayer in MoS_2 by Se sublayer or Se sublayer by S sublayer.^{37,38}

Following the success of 2D TMDs, noble metals dichalcogenides have been also investigated by the worldwide research groups. For instance, Wang *et al.*³⁹ have realized the epitaxial growth of high-quality single-crystal PtSe_2 monolayer by direct selenization of Pt. Characterizations assert its semiconductor nature that is different from the metallic bulk counterpart. A combined experimental and theoretical investigation on the layer-dependent PtS_2 monolayer properties has been carried out by Zhao *et al.*⁴⁰ Results indicate a drastic band gap reduction from monolayer (1.6 eV) to bulk counterpart (0.25 eV), which can be attributed to the strong interlayer interactions of S- p_z state. As an effective method to tune the PtS_2 and PtSe_2 monolayers properties, the formation of Janus structure PtSSe monolayer have been also studied. Various investigations have explored Janus PtSSe monolayer as promising 2D candidate for thermoelectrics,⁴¹ photocatalysis for water splitting,⁴² gas sensing,^{43,44} among others.

Recently, research interests have been focused on developing multifunctional 2D materials with two or more promising functions, which requires the artificial induction of novel feature-rich properties. In this regard, the magnetism engineering in 2D materials towards spintronic applications can be mentioned as a representative example. Generally, most of 2D materials are intrinsically non-magnetic, except for various transition metal containing members.^{45,46} Therefore, inducing significant magnetism along with feature-rich electronic properties plays a key role in order to make new spintronic materials. To the best of our knowledge, the exploration of Janus PtSSe monolayer as 2D spintronic platform has no been treated well, so far. In this work, our main aim is explore efficient magnetism engineering in Janus PtSSe monolayer through lattice defects, including vacancy and doping approach. The electronic and magnetic properties of Janus PtSSe monolayer effected by defects are analyzed in details by the spin-polarized band structure and projected density of states, Bader charge analysis, spin density, and magnetic moments. It is anticipated that Pt vacancy and doping (with P and As at chalcogen sites; with Al and Ga at Pt site) conduct to significant monolayer magnetization, meanwhile the effective band gap tuning can be achieved by chalcogen vacancies and doping with Be and Mg at Pt site. Results may suggest efficient approaches to effectively

functionalize Janus PtSSe monolayer for optoelectronic and spintronic applications.

II. Computational details

On the framework of the Density Functional Theory (DFT),⁴⁷ first-principles calculations in this work are performed using the Projector Augmented Wave (PAW) method as embedded in Vienna *ab initio* simulation package (VASP).^{48,49} Electron exchange-correlation energy is addressed with the Perdew–Burke–Ernzerhof functional of the Generalized Gradient Approximation (GGA-PBE).⁵⁰ In addition, the Heyd–Scuseria–Ernzerhof (HSE06) hybrid functional⁵¹ is also considered to correct the band diagram since PBE functional generally underestimates the electronic band gap. A plane-wave energy cutoff of 500 eV is set through the calculations. The criteria of 1×10^{-6} eV and 1×10^{-2} eV \AA^{-1} are set for the energy convergence and forces acting on each atom. First Brillouin zone of pure PtSSe monolayer is sampled with a k -mesh of $20 \times 20 \times 1$, which is generated according to Monkhorst–Pack method.⁵² In order to investigate effects of vacancy defects and doping, a $4 \times 4 \times 1$ supercell is generated, whose first Brillouin zone is integrated with a Monkhorst–Pack $4 \times 4 \times 1$ k -grid. In all cases, a vacuum layer with thickness larger than 14 \AA is added in the vertical direction to avoid the atomic interlayer interactions. The charge transfer is studied using the Bader charge analysis.^{53,54}

III. Results and discussion

A. Pristine PtSSe monolayer

The PtSSe monolayer structure is formed by sandwiching atomic S–Pt–Se sublayers in the vertical direction. Fig. 1a shows a unit cell that contains one PtSSe formula unit with three atoms (one Pt atom, one S atom, and one Se atom). Before

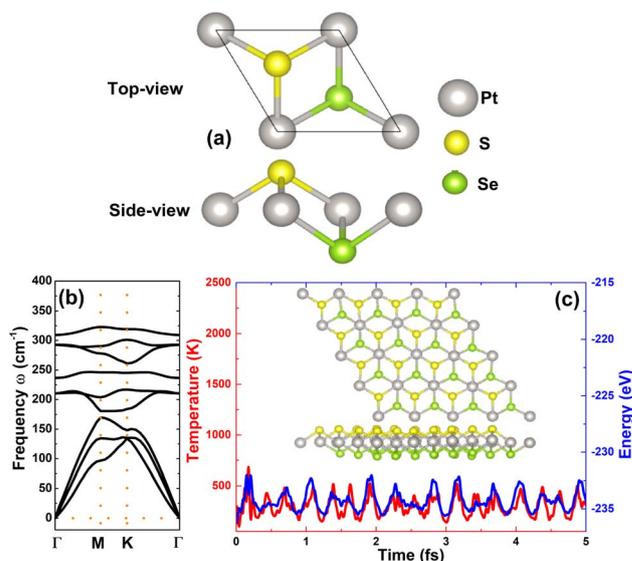


Fig. 1 (a) Optimized atomic structure, (b) phonon dispersion curves, and (c) AIMD simulations at 300 K of PtSSe monolayer.



investigating its properties, it is required the lattice optimization and the verification of stability. According to our simulations, the optimized PtSse monolayer unit cell can be described by following parameters: (1) lattice constant $a = 3.66 \text{ \AA}$; (2) chemical bond lengths $d_{\text{Pt-S}} = 2.43 \text{ \AA}$ and $d_{\text{Pt-Se}} = 2.50 \text{ \AA}$; (3) Buckling height $\Delta_{\text{total}} = \Delta_{\text{S-Pt}} + \Delta_{\text{Pt-Se}} = 1.20 + 1.34 = 2.54 \text{ \AA}$; and (3) interatomic angles $\angle \text{PtSpt} = 97.67^\circ$, $\angle \text{SPtSe} = 84.07^\circ$, and $\angle \text{PtSePt} = 94.13^\circ$. Then, the stability is analyzed as follow:

- The mechanical stability is verified using Born criteria: $C_{11} > 0$ and $C_{11} > |C_{12}|$.⁵⁵ Note that the elasticity of hexagonal 2D structure is described by only two constants C_{11} and C_{12} . Our calculations yield values $C_{11} = 78.50 \text{ N m}^{-1}$ and $C_{12} = 20.26 \text{ N m}^{-1}$. These results indicate that PtSse monolayer is mechanically stable since they satisfy the Born criteria.

- Phonon dispersion curves are displayed in Fig. 1b to examine the dynamical stability. Herein, the curves are acquired on the basis of DFPT method by applying small finite displacement in a $4 \times 4 \times 1$ supercell, using PHONOPY code.⁵⁶ From the figure, it can be concluded that PtSse monolayer is dynamically stable considering the absence of nonphysical imaginary frequency.

- *Ab initio* molecular dynamic (AIMD) simulations are conducted at room temperature (300 K) during 5 fs of time using Nose–Hoover thermostat to check the thermal stability. From results given in Fig. 1c, it can be noted that the constituent Pt, S, and Se atoms vibrates slightly around their equilibrium. None chemical bond is broken and the initial structural configuration is well preserved. Therefore, it can be confirmed that PtSse monolayer is thermally stable.

Fig. 2a shows the PtSse monolayer band structure calculated with PBE and HSE06 functional. Both functionals assert the indirect gap semiconductor character of this 2D materials considering that the valence band maximum is located along ΓK path and conduction band minimum is found between M and Γ points. PBE-based calculations provide a band gap of 1.52 eV for PtSse monolayer, which is in good agreement with previous studies.⁵⁷ It have been demonstrated that the inclusion of certain fraction of the exact Hartree exchange potential may improve the band gap calculation. Herein, the HSE06-based calculations – with 25% fraction of the exact exchange potential – yield energy gap of 2.31 eV, which is 52% larger than that obtained by PBE functional. The spectra of projected density of states (PDOS) given in Fig. 2b provide important information about the PtSse monolayer electronic structure from the orbital-decomposed point of view. Note that in the considered energy range, there is a strong electronic hybridization between Pt-5d, S-3p, and Se-4p orbitals that originate the electronic structure. Specifically, the upper part of valence band is formed mainly by Pt- d_z , S- p_z , and Se- $p_{x,y}$ states. Meanwhile, Pt- d_{xz} , S- p_z , and Se- $p_{x,y}$ states construct mainly the lower part of conduction band. The electronic hybridization may suggest significant covalent character of Pt–S and Pt–Se chemical bonds. The illustration of electron localization given in Fig. 2c shows the presence of charge in the bridge regions to confirm the covalent chemical bonds. However, the ionic character should not be neglected in these bonds because of the difference in electronegativity. The Bader charge analysis indicates that Pt atom transfers a charge

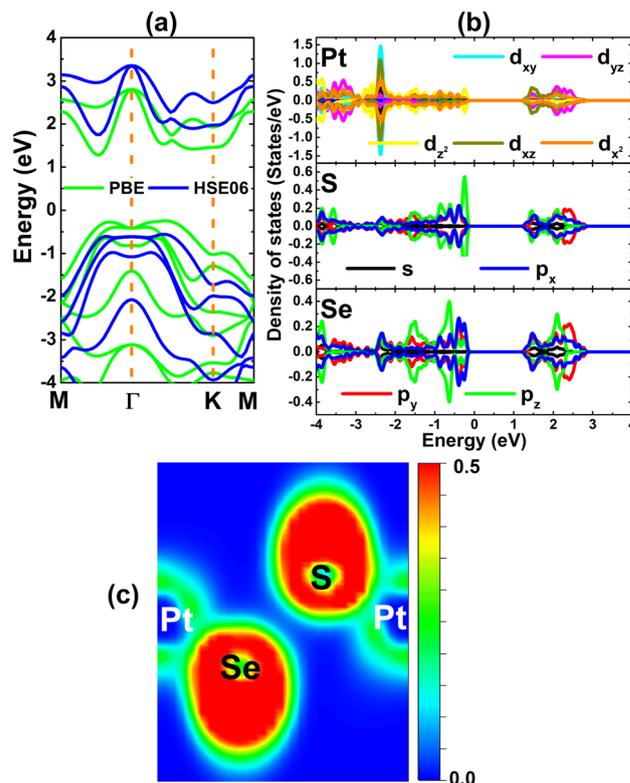


Fig. 2 (a) Electronic band structure (the Fermi level is set to 0 eV), (b) projected density of states, and (c) electron localization function of PtSse monolayer.

amount of 0.23 e to S atom, meanwhile Se atoms does not receive charge from Pt atom.

B. Effects of vacancy

In this part, the effects of vacancy on the Janus PtSse monolayer electronic and magnetic properties are investigated. Herein, single Pt vacancy (V_{Pt}), single S vacancy (V_{S}), and single Se vacancy (V_{Se}) are considered. Firstly, the formation energy E_f of defected systems is calculated as follows:

$$E_f = E_{m-x} - E_m + \mu_x \quad (1)$$

where E_{m-x} and E_m denote total energy of the defected (with a single X vacancy) and perfect monolayer, respectively; μ_x is chemical potential of X atom calculated from its most stable bulk phase ($X = \text{Pt}, \text{S}, \text{and Se}$). Then, the cohesive energy E_c is calculated to examine the structural-chemical stability using following equation:

$$E_c = \frac{E_{m-x} - [n_{\text{Pt}}E(\text{Pt}) + n_{\text{S}}E(\text{S}) + n_{\text{Se}}E(\text{Se})]}{n_{\text{Pt}} + n_{\text{S}} + n_{\text{Se}}} \quad (2)$$

herein, n_x and $E(X)$ are number of atom X in the system and energy of an isolated atom X, respectively. Results are given in Table 1. According to our computation, E_f values of 4.59, 2.06, and 1.86 eV are obtained for V_{Pt} , V_{Se} , and V_{S} system, respectively. Note that the creation of single Pt vacancy is thermodynamically less favorable since the process requires the supply of



Table 1 Formation/doping energy E_f/E_d (eV), cohesive energy E_c (eV per atom), electronic band gap E_g (eV; spin-up/spin-down; M: metallic), charge transferred from impurity (e; "−": charge gaining; "+": charge losing) and total magnetic moment M_t (μ_B) of the defected/doped PtSSe monolayer

	E_d	E_c	E_g	ΔQ	M_t
V _{Pt}	4.59	−4.23	1.27/M	—	4.00
V _S	2.06	−4.34	0.50/0.50	—	0.00
V _{Se}	1.86	−4.36	0.78/0.78	—	0.00
Li _{Pt}	1.30	−4.24	M/0.38	+0.84	1.00
Na _{Pt}	1.36	−4.23	M/0.33	+0.79	1.00
Be _{Pt}	0.88	−4.29	1.37/0.34	+1.56	2.00
Mg _{Pt}	−0.48	−4.27	1.37/0.26	+1.56	2.00
P _S	0.75	−4.45	1.22/0.54	+0.33	1.00
P _{Se}	0.68	−4.36	1.29/0.55	+0.36	1.00
As _S	0.77	−4.34	1.22/0.23	+0.44	0.99
As _{Se}	0.90	−4.35	1.39/M	+0.37	0.99

largest energy. Meanwhile, creating single Se vacancy exhibits the thermodynamic favorability over S vacancy. However, all the defected V_{Pt}, V_S, and V_{Se} systems are structural-chemically stable considering their negative cohesive energy between −4.36 and −4.23 eV per atom. In other words, the creation of single vacancies affects negligibly the system stability since E_c exhibits only slight variation (from −4.36 eV per atom of the perfect monolayer).

Fig. 3 shows the spin-polarized band structure of the defect PtSSe structures. It can be noted the appearance of new middle-gap energy branches in all cases. Single Pt vacancy induces significant spin polarization caused mostly by the middle-gap

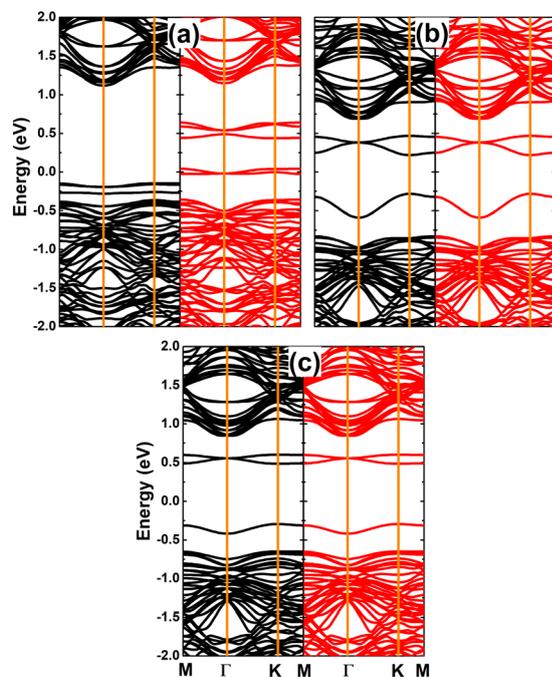


Fig. 3 Spin-resolved band structure (the Fermi level is set to 0 eV; spin-up: black curves; spin-down: red curves) of PtSSe monolayer with (a) single Pt vacancy, (b) single S vacancy, and (c) single Se vacancy.

states around the Fermi level, meanwhile the band structure is quite spin-symmetric in the regions away from the Fermi level. The band structure shows the semiconductor spin-up states with a band gap of 1.27 eV, whereas the spin-down state is metallized. Therefore, V_{Pt} structure can be classified as a half-metallic 2D material, which is expected to hold promise for spintronic applications.⁵⁸ In contrast, non spin polarization is observed in the band structures of V_S and V_{Se} systems. In these cases, new middle-gaps above and below the Fermi level leads to significant reduction of PtSSe monolayer band gap. Specifically, energy gaps of 0.50 and 0.78 eV are obtained for V_S and V_{Se} system, respectively. These results reveal the band gap reduction of the order of 67.11% and 48.68% in PtSSe monolayer induced by single S vacancy and single Se vacancy, respectively.

It is found that PtSSe monolayer is significantly magnetized upon creating single Pt vacancy, which leads to the spin-polarized band structure profile as analyzed above. According to our calculations, a total magnetic moment of 4.00 μ_B is obtained. The spin density illustrated in Fig. 4 unravels that the V_{Pt} system magnetism is originated mainly from S and Se atoms closest to the defect site. In contrast, zero magnetic moment is obtained for V_S and V_{Se} systems, indicating that the non magnetic nature is preserved upon creating single S and Se vacancies.

To investigate with more details the band structure formation and origin of magnetism, PDOS spectra of atoms around the vacancy sites are given in Fig. 5. Note that the middle-gap states of V_{Pt} system are derived mainly from S- $p_{x,y,z}$ and Se- $p_{x,y,z}$ states, which play a key role on producing the system magnetism considering their strong spin polarization. Meanwhile, Pt- d_{xz} state originates mainly the middle gap states of V_S and V_{Se} state, where important contribution from Pt- d_{xy} state to that above the Fermi is also noted. Therefore, it can be attributed the band gap reduction to Pt- d_{xy} and Pt- d_{xz} states.

The effects of vacancy concentration are also studied by increasing the number of vacancy. Herein, two, three, and four vacancies correspond to the concentration of 12.5%, 18.75%,

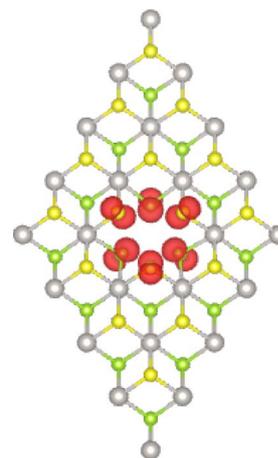


Fig. 4 Spin density (iso-surface value: 0.01 e \AA^{-3}) in PtSSe monolayer with a single Pt vacancy.



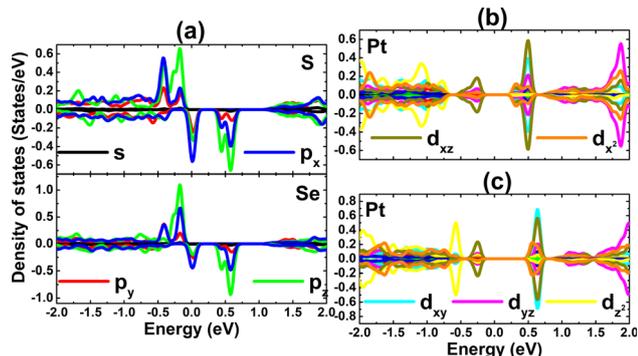


Fig. 5 Projected density of states of atoms closest to the defect site in PtSSe monolayer with (a) single Pt vacancy, (b) single S vacancy, and (c) single Se vacancy.

and 25%, respectively. Results indicate that the energy gap decreases slightly according to increase the concentration of S and Se vacancies. Otherwise, the half-metallicity is preserved upon creating 12.5% of Pt vacancy, where the magnetic properties are produced mainly by S atoms around defect sites. Interestingly, the monolayer returns to be non-magnetic semiconductor with band gap of 0.14 and 0.20 eV under effects of 18.75% and 25% of Pt vacancy, respectively (see results given in Fig. S1–S4 of the ESI† file).

C. Effects of doping at Pt sublattice

Herein, the effects of doping at Pt sublattice with alkali (Li–Li_{Pt} and Na–Na_{Pt}) and alkaline earth (Be–Be_{Pt} and Mg–Mg_{Pt}) metals on the PtSSe monolayer electronic and magnetic properties are investigated. Firstly, doping energy E_d is calculated using following formula:

$$E_f = E_{m-Pt+D} - E_m + \mu_{Pt} - \mu_D \quad (3)$$

where E_{m-Pt+D} is total energy of the doped system ($D = \text{Li, Na, Be, and Mg}$). Results are listed in Table 1. It appears that Mg doping is an exothermic process as suggested by its negative E_d value of -0.48 eV. Meanwhile, Li, Na, and Be doping requires supplying additional energy of 1.30, 1.36, and 0.88 eV, respectively. Note that doping with alkali metals is energetically less favorable than doping with alkaline earth metals considering the larger E_d values. Then, the structural-chemical stability of the doped PtSSe structures is verified by cohesive energy E_c calculated as follows:

$$E_c = \frac{E_{m-Pt+D} - [n_{Pt}E(\text{Pt}) + n_S E(\text{S}) + n_{Se}E(\text{Se}) + n_D E(\text{D})]}{n_{Pt} + n_S + n_{Se} + n_D} \quad (4)$$

From Table 1, one can see negative E_c values between -4.29 and -4.23 eV per atom, suggesting good stability of the doped systems. In other words, doping with alkali and alkaline earth metals preserves the PtSSe monolayer structural-chemical stability without any decomposition in the monolayer.

The spin-polarized band structures of the doped PtSSe Janus structures are displayed in Fig. 6. It can be noted clearly the spin

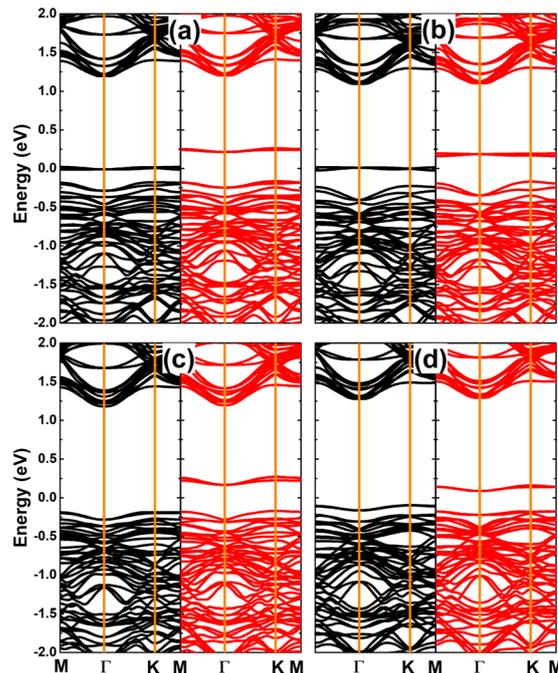


Fig. 6 Spin-resolved band structure (the Fermi level is set to 0 eV; spin-up: black curves; spin-down: red curves) of (a) Li_{Pt}, (b) Na_{Pt}, (c) Be_{Pt}, and (d) Mg_{Pt} system.

polarization at the vicinity of the Fermi level in all cases, which is caused by the appearance of new middle-gap states to determine the electronic nature. Specifically, the PtSSe monolayer spin-up state is metallized by doping with Li and Na atoms. Meanwhile, spin-down state preserves the semiconductor character with band gap of 0.38 and 0.33 eV for Li_{Pt} and Na_{Pt} system, respectively. Consequently, these doped systems can be classified as 2D half-metallic materials with a perfect spin polarization at the Fermi level. On the other hand, the diluted magnetic semiconductor nature is induced in PtSSe monolayer by doping with Be and Mg atoms, where both spin states exhibit the semiconductor character. Our calculations yield the spin-up/spin-down energy gap of 1.37/0.34 and 1.37/0.26 eV for Be_{Pt} and Mg_{Pt} system, respectively. These feature-rich electronic natures suggest the Li, Na, Be, and Mg doping as effective approaches to induce d^0 magnetism in PtSSe monolayer in order to make new 2D spintronic materials.³⁹ The Bader charge analysis indicates that metal impurities act as charge loser when they are incorporated in PtSSe monolayer. Specifically, Li, Na, Be, and Mg dopants transfer charge amount of 0.84, 0.79, 1.56, and 1.56 e to the host monolayer, respectively. This process is derived from the less electronegative nature of metal impurities than their surrounding S and Se atoms.

Our spin-polarized calculations yield total magnetic moments of 1.00 and 2.00 μ_B in PtSSe monolayer doped with alkali metals and alkaline earth metals, respectively. These results imply significant magnetization of this 2D material reached by proper doping, such that the spin polarization is produced in the band structures as analyzed above. The atom



contribution to magnetism is unraveled by the spin density illustrated in Fig. 7. From the figure, it can be noted that Se atoms closest to the doping site produces mainly magnetic properties of the doped PtSse structures.

Fig. 8 shows PDOS spectra of metal impurities and their neighboring S and Se atoms. Note that the contribution of Li, Na, Be, and Mg atoms is considerably smaller than that of S and Se atoms, where their s- $p_{x,y,z}$ states are observed. Most importantly, the middle-gap are built mainly by p_x and p_z states of S and Se atoms, which regulate the ground-state electronic behavior. The spin-polarized profiles suggest that Se- p_z state originates mainly the magnetism in the doped PtSse structures, where small contribution to magnetism may come also from S- p_x state. Weak electronic hybridization and charge transfer process may testify the ionic chemical bonds D-S and D-Se (D = Li, Na, Be, and Mg). This feature is confirmed by the electron localization function, which shows negligible electron function in the region between metal impurities and their neighboring S and Se atoms (see Fig. S5 of the ESI† file).

D. Effects of doping at S and Se sublattices

Now, doping with VA-group (P and As) atoms at S (P_S and As_S) and Se (P_{Se} and As_{Se}) sublattices is investigated. Firstly, doping

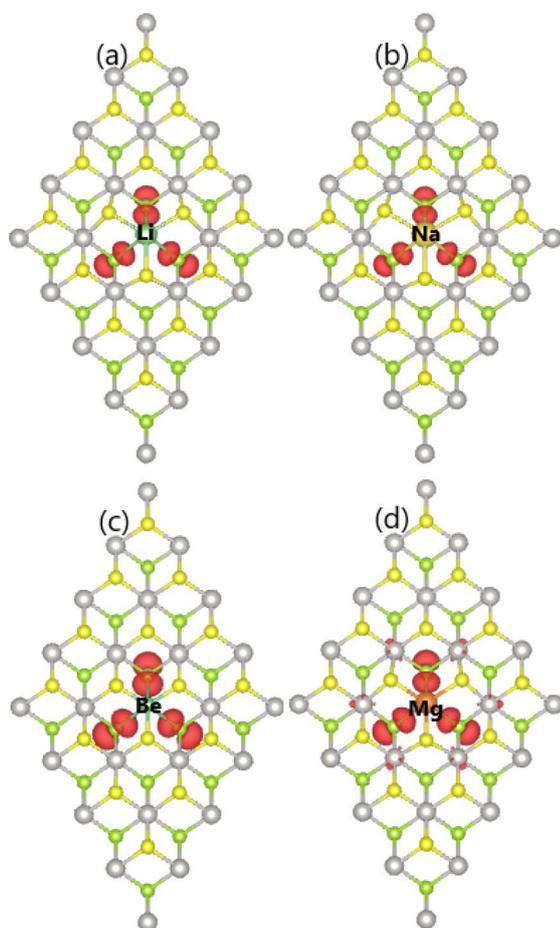


Fig. 7 Spin density (iso-surface value: $0.005 \text{ e } \text{Å}^{-3}$) in (a) LiPt , (b) NaPt , (c) BePt , and (d) MgPt system.

energy E_d and cohesive energy E_c are calculated using eqn (3) and (4) modified for these cases as follow:

$$E_f = E_{m-\text{Ch}+\text{D}} - E_m + \mu_{\text{Ch}} - \mu_{\text{D}} \quad (5)$$

and

$$E_c = \frac{E_{m-\text{Ch}+\text{D}} - [n_{\text{Pt}}E(\text{Pt}) + n_{\text{S}}E(\text{S}) + n_{\text{Se}}E(\text{Se}) + n_{\text{D}}E(\text{D})]}{n_{\text{Pt}} + n_{\text{S}} + n_{\text{Se}} + n_{\text{D}}} \quad (6)$$

where Ch = S and Se; and D = P and As. Results listed in Table 1 suggest that doping with P atom at Se site is energetically most favorable with a smallest E_d value of 0.68 eV, meanwhile largest E_d of 0.90 eV should be supplied to realize the As doping at Se site. Once formed, all the doped systems hold good structural-chemical stability as suggested by negative E_c values between -4.45 and -4.34 eV per atom, which are quite similar to that of pure monolayer (-4.36 eV per atom).

The spin-polarized band structures displayed in Fig. 9 show new features different from that of pure monolayer, which are derived from new middle-gap flat energy branches around the Fermi level. Specifically, one new state below the spin-up Fermi level and the other above the spin-down Fermi level of P_S , P_{Se} , and As_S systems induce strong spin polarization at the vicinity of the Fermi level. Consequently, these systems can be classified as 2D diluted magnetic semiconductor materials with the spin-up/spin-down band gaps of 1.22/0.54, 1.29/0.55, and 1.22/0.23 eV, respectively. Dissimilarly, the spin-down state is metallized when doping PtSse monolayer with As atom at Se site, such that the half-metallicity emerges in As_{Se} system. In this case, spin-up state exhibits the semiconductor character with a band gap of 1.39 eV. Further, the Bader charge analysis is carried out to analyze the interactions between P and As impurities with the host monolayer. Interestingly, P and As impurities act as charge losers transferring charge quantity of 0.33/0.36 and 0.44/0.37 e to the host monolayer when substituting S/Se atom, respectively. This feature is derived from the slightly less electronegative nature of P and As in comparison with the surrounding Pt atoms.⁶⁰ However, the calculated electron localization functional suggests that P–Pt and As–Pt chemical bonds are predominantly covalent (see Fig. S6 of the ESI† file), which is derived from the electronic hybridization as analyzed below.

The spin polarization in the band structures implies significant magnetization of PtSse monolayer. It is found a total magnetic moment of $1.00 \mu_B$ when doping with P atom at both S and Se sites, while As_S and As_{Se} systems have total magnetic moment of $0.99 \mu_B$. Therefore, doping with VA-group atoms can be introduced as an effective method to induce magnetism in PtSse monolayer. Consequently, new half-metallic and diluted magnetic semiconductor systems are successfully developed for spintronic applications. Furthermore, the magnetization is also reflected in the spin density surface illustrated in Fig. 10. Note that P and As impurities produce mainly the magnetic properties of P_S , P_{Se} , and As_S systems. Meanwhile, the As_{Se} system magnetism is originated mainly from the second neighbor S atoms from doping site. Considering the thermodynamic favorability of P doping at Se site in comparison with other



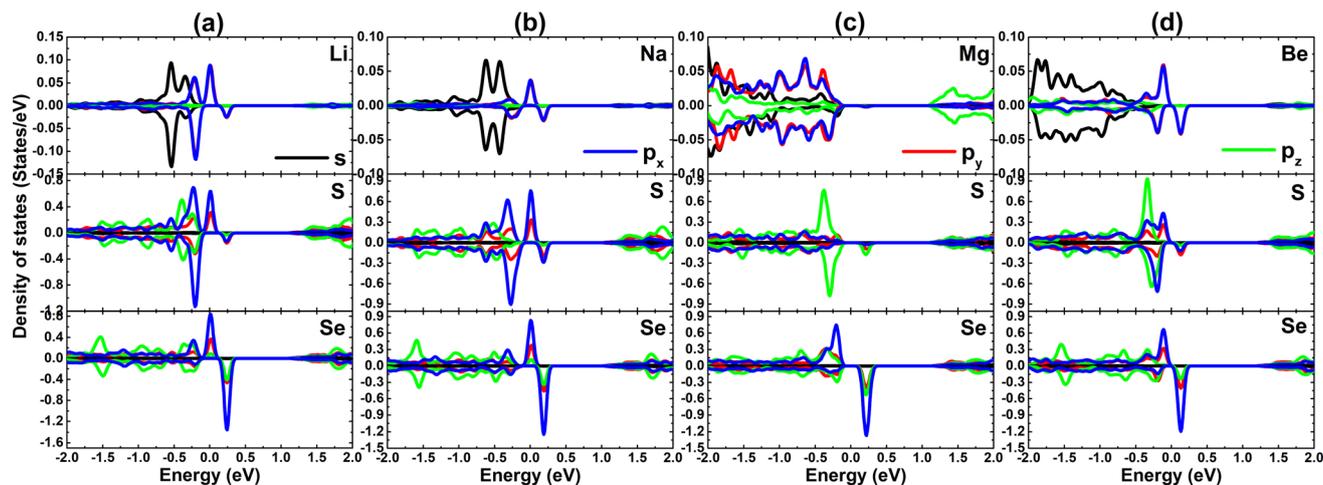


Fig. 8 Projected density of states of impurity and its neighboring S/Se atoms in (a) Li_{Pt} , (b) Na_{Pt} , (c) Be_{Pt} and (d) Mg_{Pt} system.

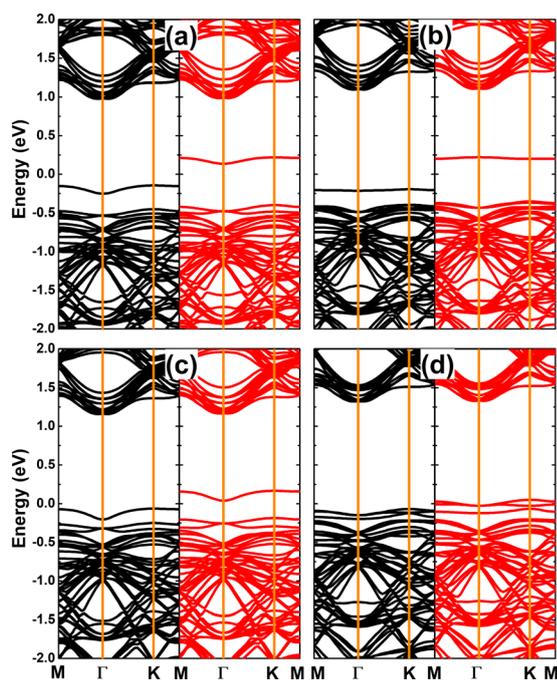


Fig. 9 Spin-resolved band structure (the Fermi level is set to 0 eV; spin-up: black curves; spin-down: red curves) of (a) P_{S} , (b) P_{Se} , (c) As_{S} and (d) As_{Se} system.

dopant atoms, the spin ordering in P_{Se} system is further studied. It is found that the antiferromagnetic \rightarrow non-magnetic \rightarrow ferromagnetic transition takes place according to increase the distance between P impurity atoms (see Fig. S7 and S8 of the ESI† file for the spin density and spin-polarized band structure, respectively).

PDOS spectra of VA-group impurities and their surrounding Pt atoms are given in Fig. 11. Note that the middle-gap states are derived mainly from P- p_z and As- p_z states with small contribution of Pt- d_{xz} and Pt- d_{z^2} states. These states exhibit strong spin polarization around the Fermi level. Therefore, one can

conclude that the magnetism is originated mainly from p_z state of VA-group impurities, where small contribution may come also from the unbalanced charge distribution between spin channels of Pt-d orbital. In the case of As_{Se} system, it can be

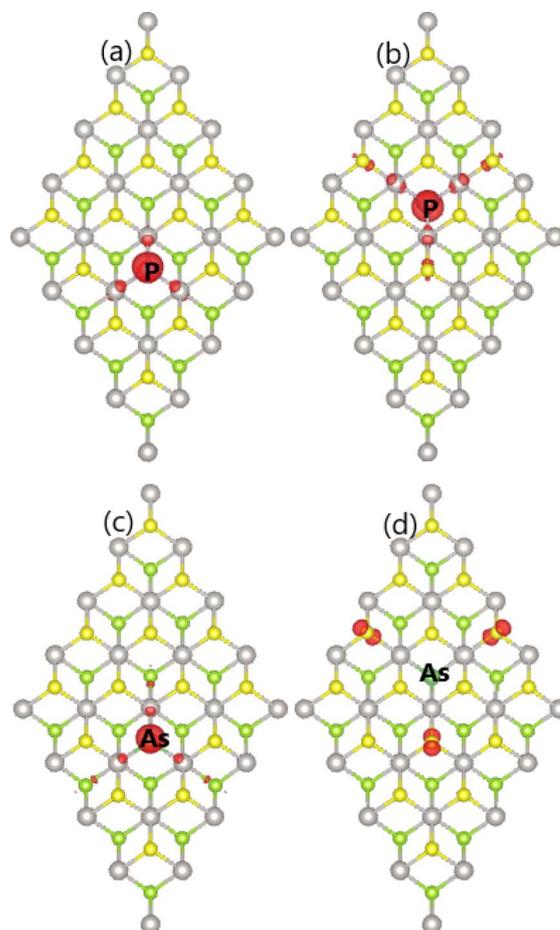


Fig. 10 Spin density (iso-surface value: $0.005 \text{ e}^{-3} \text{ \AA}^{-3}$) in (a) P_{S} , (b) P_{Se} , (c) As_{S} and (d) As_{Se} system.



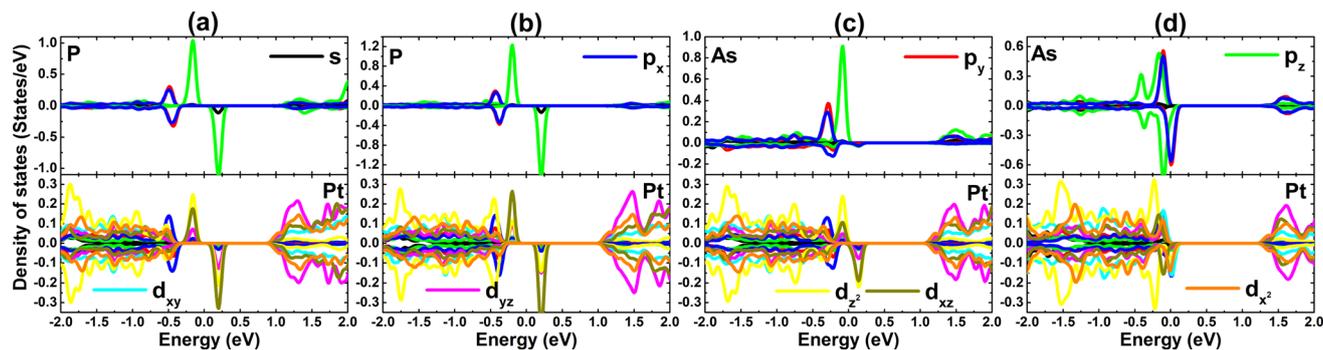


Fig. 11 Projected density of states of impurity and its neighboring Pt atom in (a) P_S , (b) P_{Se} , (c) As_S , and (d) As_{Se} system.

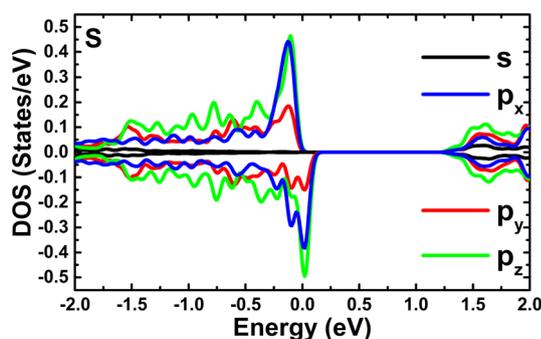


Fig. 12 Projected density of states of the magnetic S atoms in Se_{As} system.

noted also important contribution of As- p_x and As- p_y states around the Fermi level. To investigate with more details the origin of As_{Se} system magnetism, PDOS spectra of its magnetic S atoms are given in Fig. 12. From the figure, it can be attributed the appearance of magnetism to S- p_x and S- p_z states.

IV. Conclusions

In summary, the effects of vacancy and doping on the PtS_{Se} monolayer electronic and magnetic properties have been systematically investigated using first-principles calculations. Pristine monolayer possesses good mechanical, dynamical, and thermal stability. This is a 2D semiconductor materials with a relative large band gap, whose chemical bonds are predominantly covalent generated by the strong electronic hybridization between Pt-5d, S-3p, and Se-4p orbitals. By creating vacancy and realizing doping process, new middle-gap states are developed in PtS_{Se} monolayer to determine the ground-state electronic and magnetic properties. Our calculations assert that single S vacancy and Se vacancy causes considerable band gap reduction to make possible harvesting a wider light regime, which can be attributed to a larger charge distribution in 5d orbital of Pt atoms around the vacancy sites. Meanwhile, the monolayer is significantly magnetized by single Pt vacancy, doping with alkali metals (Li and Na atoms), doping with alkaline earth metals (Be and Mg atoms), and doping with VA-group atoms (P and As atoms). It has been found that V_{Pt} , Li_{Pt} , Na_{Pt} , and As_{Se} are 2D

half-metallic systems, while the diluted magnetic semiconductor nature is induced in Be_{Pt} , Mg_{Pt} , P_S , P_{Se} , and As_S systems, respectively. In all cases, impurities and those atoms closest to the defect/doping sites produce mainly the system magnetic properties. In addition, Bader charge analysis indicates that the host monolayer attracts charge from impurities. Moreover, negative cohesive energies suggest good structural-chemical stability all the defect/doped systems. Our work may introduce new multifunctional 2D materials promising for optoelectronic and spintronic applications, which are made from a non-magnetic PtS_{Se} monolayer.

Data availability

Data related to this study are available upon reasonable request.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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