


Cite this: *RSC Adv.*, 2024, 14, 15085

# Effect of calcination temperature induced structural modifications on the photocatalytic efficacy of Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> nanostructures: mechanochemical synthesis

Maqsoom Zain,<sup>a</sup> Khawaja Ansar Yasin,<sup>\*a</sup> Sirajul Haq,<sup>ID</sup> <sup>\*a</sup> Wajid Rehman,<sup>ID</sup> <sup>b</sup> Salah Ud Din,<sup>a</sup> Shafia Shujaat,<sup>a</sup> Asad Syed,<sup>ID</sup> <sup>c</sup> M. Khalid Hossain,<sup>ID</sup> <sup>d</sup> Bilal Ahamad Paray,<sup>e</sup> Jamoliddin Razzokov<sup>fg</sup> and Abdus Samad<sup>i</sup>

Water contamination due to organic pollutants is a challenging issue around the globe, and several attempts have been made to deal with this issue. Out of which, the semiconductor-based photocatalytic process had gained much attention and proved to be an efficient, easy, and economical process for the removal of organic dyes from aqueous solutions. For this purpose, the iron oxide–zirconium dioxide nanocomposite (Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NC) was prepared via a simple mechanochemical process using a mortar and pestle, followed by a calcination process at 300, 600, and 900 °C. Different physicochemical analyses were carried out in order to investigate the successful synthesis of Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NC and the effect of temperature on the crystallinity, surface area, pore size, phase composition, sample morphology, and particle/crystallite size. The Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs were subjected to a photocatalytic test under solar light irradiation against fluorescein dye in an aqueous medium, and the photocatalytic performance was examined under the influence of calcination temperatures, pH, catalyst dose, and initial concentration. The stability of the Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs was also checked by recycling them for five reuse cycles.

Received 13th March 2024  
Accepted 2nd May 2024

DOI: 10.1039/d4ra01944j

rsc.li/rsc-advances

## 1. Introduction

The occurrence of environmental pollution is marked by harm to natural elements due to the presence of detrimental synthetic substances. These substances have the potential to disturb ecosystem balance and pose diverse health risks to both animals and humans. Environmental pollution is triggered by the introduction of harmful materials, such as gaseous pollutants, toxic metals, and particulate matter (PM) into the

atmosphere; sewage, industrial effluents, agricultural runoffs, and electronic wastes into water bodies; and activities such as mining, deforestation, landfills, and illegal dumping of refuse that cause soil pollution.<sup>1</sup> Nowadays, water pollution has emerged as a widespread global issue.<sup>2</sup> Organic dyes present an environmental challenge for aquatic ecosystems.<sup>3</sup> The principal diseases linked to water pollution are acute and chronic gastrointestinal diseases, most importantly diarrheal diseases (70% of deaths attributed to water pollution), typhoid fever (8%), paratyphoid fever (20%), and lower respiratory tract infections (2%).<sup>4</sup> Fluorescein (C<sub>20</sub>H<sub>12</sub>O<sub>5</sub>) is an organic dye with low solubility in both water and other organic solvents. When dissolved, it exhibits strong fluorescence under UV light. This dye is employed in various applications such as fluid tracing, marker highlighting, fluorescent toys, and leakage detection. The toxicity of fluorescein presents a significant danger, often resulting in anaphylaxis marked by symptoms such as hypotension, tachycardia, bronchospasm, hives, and itching.<sup>5</sup>

Water containing organic dyes poses various health risks, depending on dye properties and concentrations. Potential effects include toxicity; carcinogenicity; allergic reactions; endocrine disruption, headaches, and dizziness. Numerous techniques have been identified for eliminating organic dyes from wastewater. These methods encompass equalization,

<sup>a</sup>Department of Chemistry, University of Azad Jammu and Kashmir, Muzaffarabad 13100, Pakistan. E-mail: cii\_raj@yahoo.com

<sup>b</sup>Department of Chemistry, Hazara University, Mansehra, Pakistan

<sup>c</sup>Department of Botany and Microbiology, College of Science, King Saud University, P. O. 2455, Riyadh 11451, Saudi Arabia

<sup>d</sup>Institute of Electronics, Atomic Energy Research Establishment, Bangladesh Atomic Energy Commission, Dhaka 1349, Bangladesh

<sup>e</sup>Department of Zoology, College of Science, King Saud University, PO Box 2455, Riyadh, 11451, Saudi Arabia

<sup>f</sup>Institute of Fundamental and Applied Research, National Research University TIIAME, Kori Niyoziy 39, 100000 Tashkent, Uzbekistan

<sup>g</sup>R&D Center, New Uzbekistan University, Movarounnahr Street 1, Tashkent 100007, Uzbekistan

<sup>h</sup>Department of Mechanical Engineering, Tashkent State Technical University, Tashkent 100095, Uzbekistan

<sup>i</sup>School of Material Science and Engineering, Nanjing Tech University, P. R. China



sedimentation, and various biological approaches such as bacteria-assisted, algae-assisted, fungi-assisted, yeast-assisted, and enzyme-assisted biodegradations. Additionally, chemical processes, including advanced oxidation processes (AOPs), coagulation–flocculation, and electrochemical treatments like electrocoagulation (EC), electro-fenton (EF), and anodic oxidation (AO), are employed. Physical processes, such as adsorption, membrane filtration, microfiltration (MF), ultrafiltration (UF), nanofiltration (NF), reverse osmosis (RO), ion exchange, and hybrid treatments like PMR and MBR hybrid technologies, are also utilized for this purpose.<sup>6</sup> Various methods exist for removing organic dyes from wastewater but these treatment approaches may encounter inefficiencies at times, attributed to the varied spectrum of dyes, high costs, significant sludge production, slow reaction rates, and inherent self-decomposition.<sup>7</sup>

Transition metal oxide nanoparticles as semiconductors exhibit notable characteristics, such as heightened chemical stability, environmentally friendly properties, a diverse range of band gap energies, efficient catalytic activity, and a significant surface area.<sup>8</sup> These features render them suitable for applications in heterogeneous photocatalysis.<sup>9</sup> The hematite ( $\text{Fe}_2\text{O}_3$ ) possess a band gap of 2.1–2.2 eV.<sup>10</sup> Zirconium oxide ( $\text{ZrO}_2$ ) has band gap in the 5–7 eV range.<sup>11</sup>  $\text{ZrO}_2$  holds significant importance as an oxide and has found widespread application in heterogeneous catalytic reactions.<sup>12</sup> Zirconia finds extensive applications in various fields, including bio-sensors, solid oxide fuel cells, oxygen sensors, and materials for storing  $\text{H}_2$  gas.<sup>13</sup> Iron oxide nanoparticles play a significant role in diverse applications, including magnetic and electrochemical applications, gas sensing, energy storage, cancer therapy, magnetic storage, and biomedical treatments.<sup>10</sup> The  $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$  NC had attracted significant interest due to their synergistic properties.<sup>14</sup> The  $\text{Fe}_2\text{O}_3$  offer high electrical conductivity and photocatalytic activity, while  $\text{ZrO}_2$  contribute excellent thermal stability, chemical resistance, and oxygen storage capacity. When combined at the nanoscale, these composites can exhibit enhanced electrical conductivity, improved photocatalytic performance for pollutant degradation or hydrogen generation, and potentially tunable magnetic properties.<sup>8</sup> Consequently, they have been fabricated employing various methods. IUPAC has recognized mechanochemistry as one of the ten technologies that can change the world.<sup>15</sup> Mechanochemistry provides direct reaction pathways and technologies that are solvent-free and operate at low temperatures.<sup>16</sup> Recently, there has been significant interest in utilizing mechanochemical synthesis to facilitate the production of technologically essential complex oxides.

This investigation was designed to fabricate a  $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$  nanocomposite through a straightforward mechanochemical process using a mortar and pestle, followed by calcination at various temperatures for the resulting product. The impact of calcination temperature on structural changes was analyzed using XRD, SEM, EDX, FTIR, DRS and  $\text{N}_2$  adsorption techniques. Subsequently, the  $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$  samples were employed as photocatalysts for degrading fluorescein, and the impact of

calcination temperature, initial dye concentration, catalyst dose and pH was investigated.

## 2. Materials and methods

### 2.1 Reagents

To synthesize  $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$  NCs through mechanochemical means, iron(II) hydroxide and zirconium(IV) hydroxide were utilized. These chemicals of general-purpose grade were obtained from Fisher Scientific. The hydroxides were ground with a mortar and pestle to produce the necessary metal oxides. Distilled water was employed to clean the mortar and pestle and was also used as solvent for the preparation of Fluorescein solutions.

### 2.2 Synthesis of $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$ NCs

Initially, an equal amount (1 g) of each metal hydroxide precursor was taken and ground using a mortar and pestle following standard procedures. The grinding process continued for six hours, during which all physical transformations were closely observed. The initial product exhibited a granular appearance, and with extended grinding time, it transitioned into a powdered form. Three hours into the process, the sample developed a sticky consistency, which was attributed to the breakdown of hydroxide bonds and the presence of moisture. As the six-hour mark approached, the product transitioned back into a powdered form, accompanied by the formation of metal oxides.<sup>17</sup> The resulting powder was then stored in a tightly sealed sample bottle. The experiment was conducted several times to gain required quantity of the sample and also to examine how the duration of grinding influenced the progression of the reaction. Physical characteristics, such as melting point and solubility, were also examined. Finally, the finished product underwent calcination in a muffle furnace at temperatures of 300, 600, and 900 °C.

### 2.3 Instrumentation

Various physico-chemical techniques were employed to characterize the synthesized  $\text{Fe}_2\text{O}_3$ – $\text{ZrO}_2$  NCs. The crystalline characteristics were evaluated using X-ray diffraction (XRD) analysis with the Philips X'Pert model. The Debye–Scherrer equation was utilized to ascertain the crystallite size of the synthesized nanostructures. A field emission scanning electron microscope (SEM), specifically the JEOL JSM-5600LV model from Tokyo, Japan was employed to investigate the microstructure and surface topology. EDX (model INCA-200 (UK)) was employed to verify the elemental composition of the sample. The  $\text{N}_2$  adsorption experiment was conducted using the Gemini instrument model 2390 t at a standard pressure of 760.00 mmHg, and the BET equation was utilized to assess the surface area and pore size distribution. UV-DRS analysis was employed to investigate the light absorbance characteristics of the samples, and the band gap energy was determined using Tauc plot. The analysis of surface functional moieties was conducted through FTIR, spanning the range of 4000 to 400  $\text{cm}^{-1}$ . The decrease in the  $\lambda$ -max of fluorescein dye during photocatalytic



process was examined through UV-visible spectrophotometer (1601 SHIMADZU).

## 2.4 Photocatalytic assay

The photocatalytic efficiency of all the samples of prepared nanocomposite *i.e.*  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$ , utilizing mechanochemical method, including both uncalcined and those calcined at 300, 600, and 900 °C, was assessed in the context of the photocatalytic degradation of fluorescein dye. Initially, a 15 ppm stock solution of fluorescein dye was prepared in distilled water. 100 ml of this solution was transferred into a reaction vessel, to which 25 mg of the synthesized samples were added. To achieve adsorption-desorption equilibrium, the reaction mixture was stirred in the dark for 30 min. Following exposure of the solution to simulated solar radiation for a specific duration, a thorough assessment of the sample was conducted using a double-beam spectrophotometer, and the reduction in absorbance maxima was monitored over time.

## 3. Results and discussion

### 3.1 XRD analysis

The XRD patterns displayed in Fig. 1, shows three sets of corresponding to  $\text{Fe}_2\text{O}_3$ ,  $\text{ZrO}_2$  and  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$ . The diffraction bands appeared at 27.17, 40.54, 49.05, 53.92, 71.79 and 75.02 positions corresponding to (012), (113), (024), (116), (10 10) and (217). All these peaks are found similar to those reported in JCPDS card 01-079-1741 attributed to the rhombohedral crystal system with space group of  $R\bar{3}c$  and space number of 167. The length of two  $a$  and  $b$  coordinates are same (5.0342 Å),  $c$  is equal to 13.7460 Å, the interfacial angles  $\alpha$  and  $\beta$  is equal to 90.00° whereas  $\gamma$  is of 120.00°. The density and volume of unit cell is found to be 5.27 g cm<sup>-3</sup> and 301.69 × 10<sup>6</sup> pm<sup>3</sup> respectively. The second set of diffraction peaks appeared at 31.24, 34.01, 35.27, 38.50, 44.46, 49.99, 53.94, and 55.18 due to the diffraction of X-ray from the  $hkl$  planes (111), (020), (002),

(021), (211), (220), (221), and (300). These peaks are matched with those reported in reference card 96-900-5835 and assigned to the orthorhombic geometry of  $\text{ZrO}_2$  crystal system having space group of  $Pbcm$  and space number 57. The length of  $a$ ,  $b$  and  $c$  coordinates are 5.0050, 5.2350 and 50.10 Å respectively and degree of interfacial angles  $\alpha$ ,  $\beta$  and  $\gamma$  are 90.00. The density and volume of unit cell 6.18 g cm<sup>-3</sup> and 132.34 × 10<sup>6</sup> pm<sup>3</sup> respectively. The diffraction peaks along the  $hkl$  planes at 27.95 (111) and 46.36 (112) is assigned to the monoclinic geometry of  $\text{ZrO}_2$ , which is in accordance to the reported literature.<sup>18</sup>

The XRD analysis of the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs calcined at different temperatures reveals the influence of thermal treatment on the phase composition, crystallinity and crystallite size. With increasing calcination temperature, no new peak was emerge in the diffractogram confirmed that no phase transformation occurred. The increased intensity of the peaks suggest that the crystallinity of the sample increases and the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NC calcined at 900 °C has the highest degree of crystallinity. Therefore, the observed changes in the peaks intensities with increasing calcination temperature suggests that higher temperatures promote the formation of more ordered and stable crystalline  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NC.

### 3.2 SEM analysis

SEM is a powerful imaging technique offering high-resolution, three-dimensional views of material surfaces in the micro- and nanoscale. Under low magnification, the SEM image of the uncalcined  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs highlighted differences in both size and shape, revealing a rough morphology (Fig. 2(a)). Various aggregates and agglomerates with diverse morphologies were

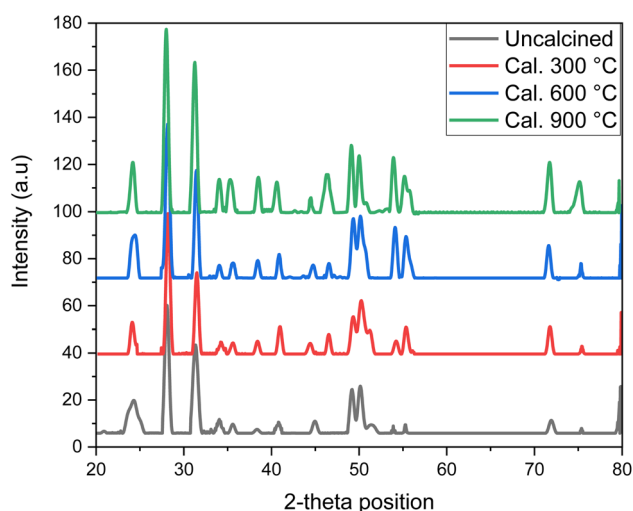


Fig. 1 The XRD patterns of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs (uncalcined and those calcined at 300, 600 and 900 °C).

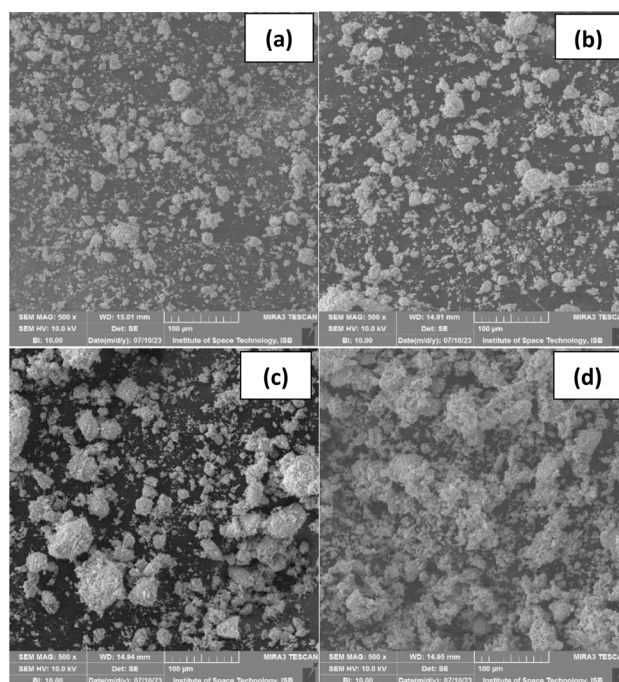


Fig. 2 Low magnification SEM images of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs (uncalcined (a) and calcined at 300 (b), 600 (c) and 900 (d) °C).



visible, some with clear boundaries. These structures, along with individual particles, were randomly dispersed, with larger spacing among them. Overall, their surfaces displayed a non-smooth appearance. After calcination at 300 °C (Fig. 2(b)), the SEM examination revealed significant variations in size and shape, emphasizing the influence of calcination temperature. There was a distinct increase in agglomeration, especially noticeable at specific points, maintaining a non-smooth surface appearance. The spacing among some agglomerates became more prominent at certain points. Following calcination at 600 °C, the SEM analysis (Fig. 2(c)) exhibited notable disparities in sizes and shapes, with a substantial increase in agglomeration. The agglomerates, with non-smooth surfaces, appeared scattered and displayed mixed morphologies. Post-calcination at 900 °C (Fig. 2(d)), pronounced agglomeration reduced spacing, presenting non-smooth surfaces with cavities visible only at few points. The resulting agglomerates showcased diverse morphologies.

It was employed to explore the impact of calcination temperature on the morphology of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs, with results presented in Fig. 3(a-d). Under high magnification, the SEM analysis of uncalcined  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs revealed variations in sizes and shapes, indicating a rough morphology and non-uniformity. The presence of agglomerates with diverse shapes, small dispersed particles, and irregularly placed smaller nanocomposites on larger agglomerates was observed. Despite a smooth surface, uneven distribution led to cavities. The rough morphology was attributed to variations in nucleation and growth. Calcination of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NC at 300 °C showed significant variations, increasing agglomeration, and reducing cavities due to fusion of particles and agglomerates. Larger

agglomerate distribution increased, with elevated scattering of small nanocomposites. At 600 °C, pronounced agglomeration, reduced cavities, and increased scattering were noted. Calcination at 900 °C led to extensive agglomeration, fusion of particles, reduced cavities, and indistinct boundaries, presenting a smooth appearance with mixed morphologies. Along with morphological changes, calcination temperature has also effect particles size and the average size for  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs are found to be 38.60, 42.97, 55.21 and 61.78 nm for the uncalcined, and those calcined at 300, 600 and 900 °C respectively. The heat treatment at higher temperature led to the diffusion and coalescence of the adjacent particles results into the formation of larger particles. Moreover, the elevated temperature promote the particles growth and agglomerations compared to nucleation process led to the formation of larger particles that influence the surface area and reactivity of the sample.

### 3.3 Surface area and pore size analysis

The BET equation was applied to  $\text{N}_2$  adsorption data to examine the surface area and pore size of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs. The BET plots are shown in Fig. 4(a-d), and the low  $1/[Q(p^\circ/p - 1)]$  value in the BET plot shows that the larger quantity of  $\text{N}_2$  was adsorbed at a lower relative pressure ( $0.5 p/p^\circ$ ). A gradual decrease was seen in the adsorption of  $\text{N}_2$ , with increasing relative pressure attributable to surface/pore saturation and gas-gas interaction. If we compared the BET plots for all the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  samples, a significant increase occurred in the  $1/[Q(p^\circ/p - 1)]$  values with the increasing calcination process, which suggests a clear decrease in the adsorption of  $\text{N}_2$  into the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  surface. This decrease is due to the calcination temperature-induced morphology changes that occurred in the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  samples, including agglomeration, fusion of particles, and reduced pore size. The  $\text{N}_2$  adsorption isotherms of the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs are shown in Fig. 5, which found similar with type IV isotherm according to IUPAC classification. The type of

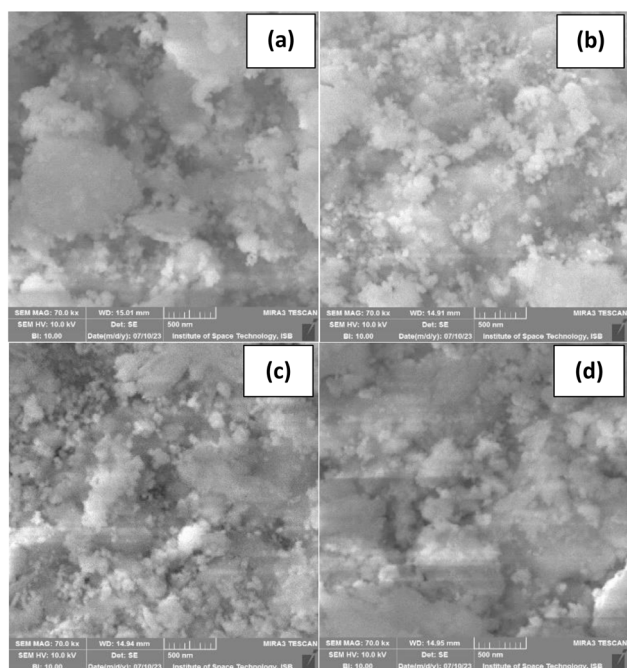


Fig. 3 High magnification SEM images of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs (uncalcined (a) and calcined at 300 (b), 600 (c) and 900 (d) °C).

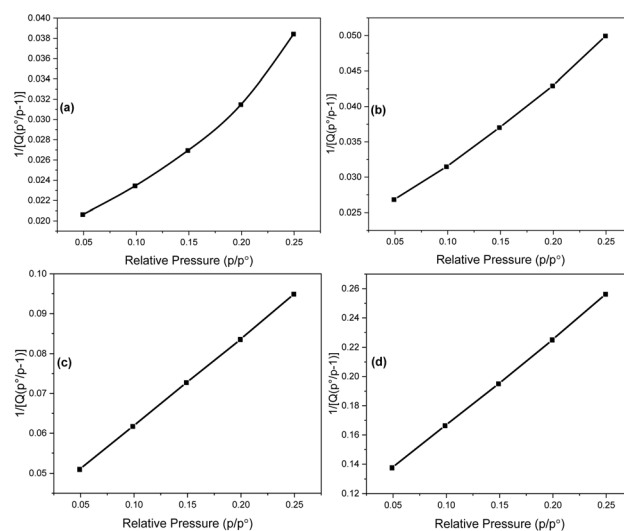


Fig. 4 BET plots showing the adsorption of  $\text{N}_2$  on the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs; uncalcined (a), cal. at 300 °C (b), cal. at 600 °C (c), and cal. at 900 °C (d).



isotherms suggest the presence of mesopores in the samples. The pore distribution plots are shown in the inset in Fig. 4 shows that the size pores decreases with increasing calcination temperature. The data listed in Table 1 which shows that the uncalcined Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NC exhibits the highest surface area as compared to the calcined analogues. A gradual decrease in the surface area of the samples is seen with increasing calcination temperature, which might be due to increasing crystallite or particle sizes, agglomeration, and other morphological changes. Similarly, the pore sizes and pore volume of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs are also decreased with increasing calcination temperature. The decrease in pore parameters is attributed to the agglomeration process, an increase in particle size, and the rearrangement of the particles that led to the collapse or blockage of pores.<sup>19</sup>

Surface area is one of the main factors that determines the photocatalytic efficacy of the materials, and the photocatalytic reaction occurs at the active sites that are present on the surface of the catalyst. Thus, catalysts with a higher surface area exhibit more active sites that facilitate the photocatalytic process. Furthermore, the catalyst with a larger surface area has a larger number of atoms on the surface, which enhances the absorption of incident light and results in the efficient generation of electron-hole pairs that initiate the redox process.<sup>20</sup>

### 3.4 DRS analysis

The light absorption phenomena of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs (uncalcined and calcined at 300, 600 and 900 °C) was studied by DRS (inset: Fig. 6(a)) and all the samples shows maximum absorption in the UV region, which is a distinctive feature of many metal oxides including Fe<sub>2</sub>O<sub>3</sub> and ZrO<sub>2</sub> due to their specific electronic band structure. A slight red shift was observed in the absorbance edge of the samples is attribute to an increase in the crystallinity with increasing calcination temperature. As the larger particles have a lower band gap,

**Table 1** Texture parameters surface area, pore size and pore volume of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs (both uncalcined and calcined at different temperature)

Samples	Texture parameters		
	$S_{\text{BET}}$ (m <sup>2</sup> g <sup>-1</sup> )	Pore size (Å)	Pore volume (cm <sup>3</sup> g <sup>-1</sup> )
Uncalcined	92	66.28	0.0377
Cal. 300 °C	64	17.23	0.0285
Cal. 600 °C	21	15.97	0.0109
Cal. 900 °C	2	15.17	0.0002

where quantum confinement effect become less dominant as the particle size increases.

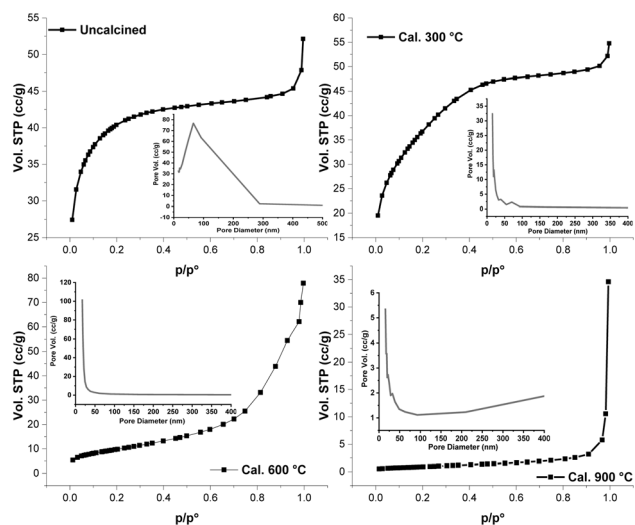
The shift toward longer wavelength may also be due to the redistribution of surface defects *i.e.* vacancies/interstitials within the crystal, they can also act as a charge carrier and disturbed the electronic band structures.<sup>21</sup> This shift is attributed a decrease in the band gap energy. The band gap energies for direct (Fig. 6(a)) and indirect (Fig. 5(b)) transitions of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs were determined through the eqn (1) and (2) respectively; where the  $\alpha$  is absorption coefficient,  $h$  is plank's constant,  $\nu$  photon frequency,  $A$  is proportionality constant and  $E_g$  is the band gap energy.<sup>22</sup> The calculated band gap energies are listed in Table 2, showing a gradual decreasing trend with increasing calcination. This decrease in the band gap energy is might be due to the increase in the crystallinity and particle size at elevated calcination temperature.<sup>23</sup> The calculated band gap energies, they are found to be less than those reported previously.<sup>24</sup> The difference in the band gap energies for both direct and indirect transitions attributed to the variation particle/crystallite size, which is in accordance with reported literature.<sup>25</sup>

$$\alpha h\nu = A(h\nu - E_g)^2 \quad (1)$$

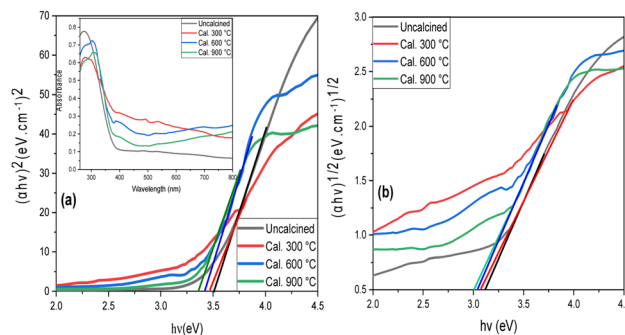
$$\alpha h\nu = A(h\nu - E_g)^{1/2} \quad (2)$$

### 3.5 EDX analysis

An EDX spectrum identifies and quantifies elements in a sample by measuring characteristic X-ray emissions, enabling



**Fig. 5** N<sub>2</sub> adsorption of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs (inset: pore size distribution); uncalcined, cal. at 300 °C, cal. at 600 °C, and cal. at 900 °C.



**Fig. 6** Tauc's plots for direct (a) and indirect (b) transition (inset: DRS spectra) of the Fe<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> NCs both uncalcined and those calcined at different temperature.



**Table 2** Band gap energies for direct and indirect transition in Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs

Band gap (eV)	Fe <sub>2</sub> O <sub>3</sub> –ZrO <sub>2</sub> NCs			
	Uncalcined	Cal. 300 °C	Cal. 600 °C	Cal. 900 °C
Direct band gap	3.52	3.45	3.40	3.35
Indirect band gap	3.12	3.08	3.04	3.01
Direct spectrum band gap (=1240/ λ)	3.25	3.19	3.14	3.10

elemental analysis, spatial mapping, and material characterization. For Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs, both uncalcined and calcined at various temperatures (300, 600, and 900 °C), the EDX spectrum unveiled the elements' presence and their relative abundance in the sample (Fig. 7(a)). The signal observed at 0.2 keV indicated the existence of oxygen. This peak was probably linked to the oxide constituents present in the nanocomposites.

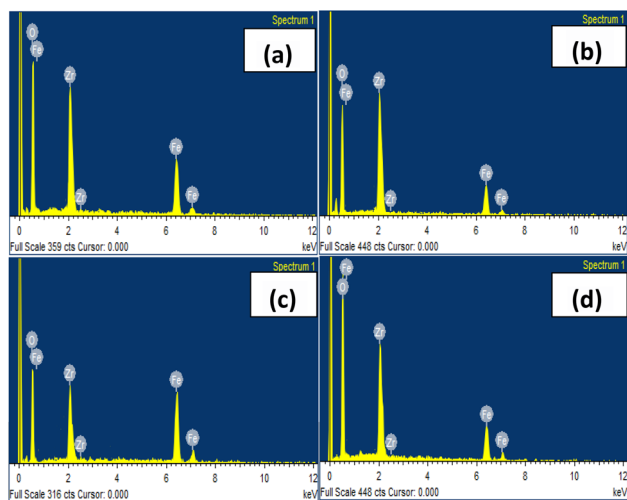
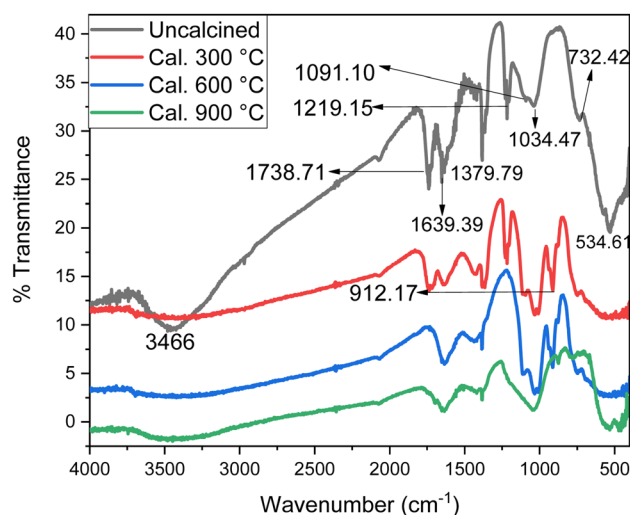
The peaks at 0.3, 6.2, and 6.5 keV unequivocally represented characteristic X-ray signals indicative of the presence of iron whereas the peaks observed at 2 and 2.2 keV suggested the presence of zirconium. Elevating the calcination temperature induces diverse effects on the EDX spectra of Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub>. In Fig. 7(b–d). The EDX spectra revealed an increase in the peak intensities of Fe and Zr with increasing calcination temperature. This variation in peak intensities for the same elements can be ascribed to multiple factors, including sample inhomogeneity, variations in sample preparation, differential absorption, surface contamination, matrix effects, crystal structure changes, detector sensitivity, and beam damage. These factors collectively contribute to the complexity of the interaction between the electron beam and the sample, resulting in fluctuations in X-ray emissions and subsequent peak intensities.

Studies additionally propose that the phase composition, morphology, and properties of materials are influenced by the calcination temperature.

In the Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> system, solubility of Fe<sub>2</sub>O<sub>3</sub> in the ZrO<sub>2</sub> lattice undergoes significant changes at different calcination temperatures, exhibiting lower solubility at higher temperatures. Furthermore, calcination temperature impacts the crystalline behavior, enhancing the crystallinity of ZrO<sub>2</sub> NPs as the temperature increases. Moreover, peak intensities of elements in EDX spectra can increase with higher calcination temperatures due to various factors, including crystallite size, phase transformation, enhanced crystallinity, coarsening of nanoparticles, and higher phase purity. Notably, different phases or crystal structures manifest distinct characteristics in terms of X-ray emissions in EDX spectra, owing to the unique atomic composition and arrangement within each phase. Each element emits specific X-rays based on energy differences between electron shells, resulting in characteristic X-ray lines in the spectra. Additionally, variations in lattice arrangements and crystal structures contribute to different diffraction patterns, influencing the shape and position of X-ray peaks in the EDX spectra. These diverse factors collectively contribute to the observed spectra.

### 3.6 FTIR analysis

The FTIR spectra of Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs shown in Fig. 8, exhibits peaks aligned with specific wavenumbers (cm<sup>−1</sup>), signifying the points where the transmission of infrared radiation took place.

**Fig. 7** EDX spectra of the Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs both uncalcined and those calcined at different temperature.**Fig. 8** FTIR spectra of the Fe<sub>2</sub>O<sub>3</sub>–ZrO<sub>2</sub> NCs both uncalcined and those calcined at different temperature.

The broad band centered at  $3466\text{ cm}^{-1}$  was assigned to the stretching collision of H–O–H and hydroxyl absorption.<sup>26</sup> Peak at  $1738.71\text{ cm}^{-1}$  was attributed to the Zr–OH functional group.<sup>27</sup> A midrange band at  $1639.39\text{ cm}^{-1}$  was noted, possibly originating from the bending of H–O–H coordinated  $\text{H}_2\text{O}$  molecule.<sup>28</sup> Peak at  $1379.79\text{ cm}^{-1}$  was ascribed to the presence of surface –OH groups in M–OH (either  $\text{Fe}(\text{OH})_2$  or  $\text{Zr}(\text{OH})_4$ ).<sup>26</sup> An additional peak emerged at  $1219.15\text{ cm}^{-1}$ , which may also be attributed to the vibrations of M–OH (either  $\text{Fe}(\text{OH})_2$  or  $\text{Zr}(\text{OH})_4$ ).<sup>29</sup> A weak band emerged at  $1091.10\text{ cm}^{-1}$  possibly arising due to terminal metal hydroxide(s).<sup>30</sup> Peak at  $1034.47\text{ cm}^{-1}$  might also have happened due to the different modes of bending vibration of the M–O bond (either Fe–O or Zr–O).<sup>31</sup> At  $912.17\text{ cm}^{-1}$ , a sharp peak was identified as the result of O–M–O vibrations (can be Fe–O–Fe, Fe–O–Zr, or Zr–O–Zr).<sup>32</sup> A midrange band at  $732.42\text{ cm}^{-1}$  occurred due to the Zr–O–Zr asymmetric stretching modes.<sup>33</sup> A sharp band was obvious at  $534.61\text{ cm}^{-1}$  due to the Fe–O functional group.<sup>27</sup> These peaks consistently appeared within the identical range throughout the spectra of all samples. For the samples that were calcined at 300, 600, and 900 °C, there was a progressive decline in the strength of the band in the  $3466\text{ cm}^{-1}$  range. In the calcined samples, there was a noticeable drop in the band's strength in the  $1738.71\text{ cm}^{-1}$  range suggest the condensation of  $\text{Zr}(\text{OH})_2$  to  $\text{ZrO}_2$ . In the  $1639.39\text{ cm}^{-1}$  range, the peaks showed the same decline. A sharp drop was seen in the intensities of the peaks at  $1639.39\text{ cm}^{-1}$  and  $1379.79\text{ cm}^{-1}$  also suggest the dehydration of the samples at elevated temperature. The increase/decrease along with a minor shift in the intensity of the bands at  $1219.15\text{ cm}^{-1}$ ,  $1091.10\text{ cm}^{-1}$ ,  $1034.47\text{ cm}^{-1}$ ,  $912.17\text{ cm}^{-1}$ , and  $732.42\text{ cm}^{-1}$  attributed to the calcination temperature promoted rearrangement of the unit cell in the crystal structure.<sup>24</sup> The band at  $534.61\text{ cm}^{-1}$  seemed to begin merging together in calcined samples; this merging effect became more noticeable at higher calcination temperatures, when it began to transform into a broad band. The reduction happened as a result of a condensation reaction that evaporated water molecules and turned metal hydroxides into metal oxides.<sup>34,35</sup> It was also ascribed to the generation of metal oxides and the increasing concentration of metal oxides, the intensity of the bands in this region was on the rise.<sup>25</sup>

### 3.7 Photocatalytic activity

The photocatalytic efficiency of  $\text{Fe}_2\text{O}_3\text{--ZrO}_2$  NCs, synthesized *via* the mechanochemical method, was evaluated for both uncalcined samples and those exposed to different calcination temperatures against fluorescein dye. The assessment took place outdoors, amid natural sunlight, spanning June 15 to 30, and occurred between 11 a.m. and 3 p.m. The diminishing greenish-yellow hue over time signified the decolorization of the reaction mixture. Following the UV-visible analysis conducted using a double-beam spectrophotometer, there was an initial noticeable decline noted in the absorbance maxima at 475 nm. This suggested the degradation of the chromophore responsible for light absorption at that particular wavelength. The degradation profile shown in Fig. 9(a) demonstrated a gradual

diminution in the absorbance maximum with the rise in calcination temperature. During the initial 20 min, especially in the uncalcined sample in comparison to those subjected to various calcination temperatures, there was a noticeable and sharp decline in the absorbance maxima. Afterward, the rate of reduction decelerated in all samples. The most significant decline in the absorbance maxima was evident in the uncalcined sample.<sup>35,36</sup>

The degradation percentage of Fluorescein was computed using eqn (1),<sup>37</sup> and the results are illustrated in Fig. 9(b) for different  $\text{Fe}_2\text{O}_3\text{--ZrO}_2$  NCs. This included both the uncalcined sample and samples exposed to different calcination temperatures, namely 300, 600, and 900 °C. The uncalcined sample showed a fluorescein degradation of 83.73%, whereas the samples subjected to calcination at 300, 600, and 900 °C showcased degradation percentages of 78.09%, 56.86%, and 43.13%, respectively. The most photocatalytically active prepared sample was the one that was left uncalcined, whereas the sample that was calcined at the maximum temperature *i.e.* 900 °C showed the lowest level of activity. The rate of the photochemical reaction was depicted by the rate constant values, which were calculated using eqn (2) and displayed in Fig. 9(c).<sup>37</sup> The photocatalytic processes carried out in the presence of uncalcined NCs and those calcined at 300, 600, and 900 °C were found to be 0.0198, 0.0133, 0.0083, and 0.0055 quantity of dye was degraded per min, respectively, based on the obtained results. These results represented the relative photocatalytic effectiveness of each prepared sample in boosting fluorescein degradation in the specified reaction conditions.<sup>38,39</sup>

A greater rate constant denoted a faster rate of degradation, indicating increased photocatalytic activity in the  $\text{Fe}_2\text{O}_3\text{--ZrO}_2$

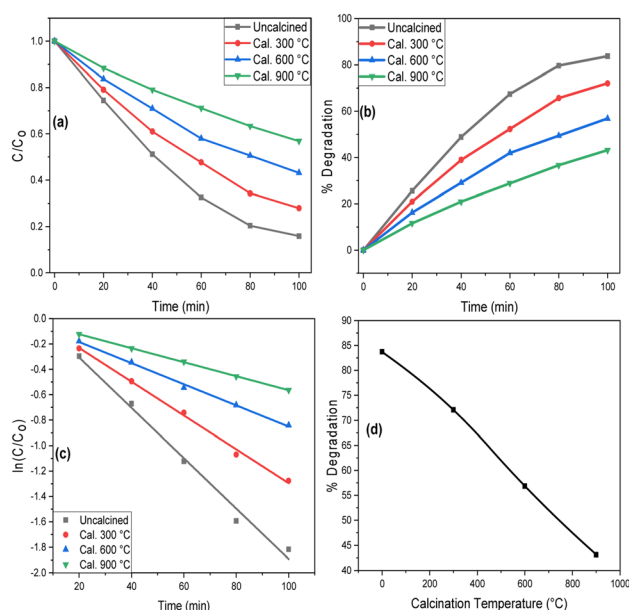


Fig. 9 The photocatalytic activity of  $\text{Fe}_2\text{O}_3\text{--ZrO}_2$  NC against fluorescein: (a) degradation profile in the presence of both light and catalyst, (b) percentage degradation, (c) rate constant, and (d) effect of calcination temperature.



that was produced. With a value of 0.0198 per min, the uncalcined NCs showed the highest rate constant among the examined samples, according to the results. Additionally, the results suggested that uncalcined NCs might be more effective in promoting the targeted compound's degradation.

$$\% \text{ Degradation} = \frac{C_o - C_e}{C_o} \times 100 \quad (3)$$

$$\ln\left(\frac{C}{C_o}\right) = -kt \quad (4)$$

The process of photodegradation for fluorescein, facilitated by a  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  catalyst under solar light, comprises a sequence of steps orchestrated by the interplay among the catalyst, sunlight, and the dye molecule Fig. 10. The catalyst absorbs photons from solar light, initiating the formation of electron-hole pairs ( $e^-/h^+$ ) within the catalyst material. Under these conditions,  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  acted as a semiconductor; photons that were absorbed raised electrons in the catalyst's valence band (VB) to the conduction band (CB), which led to the creation of electron-hole pairs. The VB of  $\text{ZrO}_2$  was where the positive holes congregated, while the VB of  $\text{Fe}_2\text{O}_3$  contained the excited electrons. These charge carriers were essential to the succeeding redox processes. Water molecules reacted with photogenerated holes to form hydroxyl radicals ( $\cdot\text{OH}$ ), whereas oxygen molecules were reduced by photogenerated electrons to become superoxide radicals ( $\cdot\text{O}_2^-$ ), which then reacted with hydrogen to produce hydroxyl radicals. The highly reactive hydroxyl radicals ( $\cdot\text{OH}$ ) attacked the adsorbed fluorescein molecules, causing the dye's chemical bonds to dissolve and the dye to mineralize into carbon dioxide and water.<sup>40</sup>

### 3.8 Determinants of photocatalytic activity

**3.8.1 Effect of pH on % degradation.** The pH of the fluorescein solution was modified with 0.1 M HCl and 0.1 M NaOH. The influence of pH fluctuations on the percentage of dye

degradation was closely observed (Fig. 11(a)). Under acidic conditions ( $\text{pH} < 6.63$ ),  $\text{ZrO}_2$  carries a positive charge, while in alkaline environments ( $\text{pH} > 7.0$ ), it bears a negative charge.<sup>41</sup> Moreover, fluorescein is found in the pH range of 4.3 to 6.4 in its monoanionic form, which changes to its dianionic form at 6.4. The results showed that 62.18, 72.31, 80.01, 83.73, and 76.29 percent of fluorescein was degraded at pH 4, 5, 6, 7 and 8 in 100 min respectively.

The data suggested that within an acidic environment, the forces of attraction between the dye and catalyst were conducive to the dye being adsorbed onto the surface of the photocatalysts. The inclination toward adsorption was ascribed to the dye being in an anionic form under acidic conditions, while the catalyst was in a cationic form. This fostered a significant interaction between them, ultimately resulting in the adsorption of the dye and consequently causing a higher percentage of degradation, reaching its optimum at a pH of 6. Furthermore, the surface of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  was found to become negatively charged at high pH values, which resulted in repulsive forces that inhibited adsorption. In other words, when the pH of the solution was raised above 7, a large proportion of the adsorbed dye molecules were not in direct contact with the photocatalysts surface, which lowered the rates of degradation.<sup>42</sup>

**3.8.2 Effect of dose on % degradation.** The degradation of fluorescein was carried out with multiple catalyst doses (10, 15, 20, 25, and 30 mg). The results showed a direct increase in percent degradation as the catalyst dose increased (Fig. 11(b)). It is found that when the concentration of the catalyst increases, the rate of degradation for the dye's breakdown increases.<sup>43</sup> Peak activity for the degradation of fluorescein dye in the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NC was detected at a catalyst dosage of 25 mg, indicating an ideal concentration for the photocatalytic process. Increasing the dose of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NC from 10 to 25 mg resulted in a rise in fluorescein degradation percentage from 43.69% to 83.73%, whereas at 30 mg, it fell to 70.86%. This was due to the NC's increased total surface area, which improved the active sites that interacted with dye.<sup>44</sup> Production of reactive radical

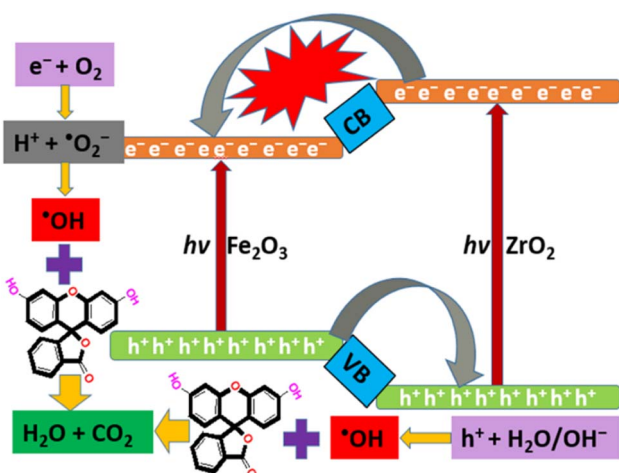


Fig. 10 Proposed photocatalytic mechanism showing the electron-hole generation and subsequent photocatalytic reactions.

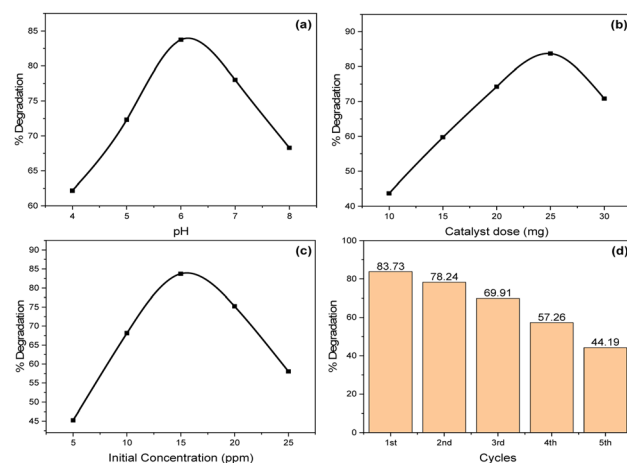


Fig. 11 The factors that affect the photocatalytic activity of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs against fluorescein: (a) effect of pH, (b) effect of catalyst dose, (c) effect initial concentration, and (d) recycling of the catalyst.





risers with the number of active sites on NCs which improves degradation. However, the development of intermediate byproducts was indicated by the decrease in dye degradation at higher doses, specifically at 30 mg. Furthermore, aggregation of NCs may happen at greater doses, potentially leading to decreased degradation. This could be caused by the catalyst absorbing light less effectively or by photons not interacting with the catalyst surface in a useful way.<sup>45</sup>

**3.8.3 Effect of initial concentration on % degradation.** Different initial concentrations (5, 10, 15, 20, and 25 ppm) were used for the photocatalytic degradation of fluorescein, yielding degradation percentages of 45.21, 68.14, 83.73, 75.22, and 58.05, respectively as shown in Fig. 11(c). As the initial dye concentration increased to 15 ppm, a rise in the percentage of dye degradation was noted. On the other hand, the % degradation was shown to decrease as the original concentration was increased further. At an initial concentration of 15 ppm, the maximum degradation efficiency (83.73%) was obtained, suggesting that this is the ideal concentration for increased photocatalytic activity.

The proportion of degradation decreased as the concentration of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs increased because their active sites were fully occupied and saturated. Increased concentrations may interfere with light-catalyst surface interaction, which could lead to reduced hydroxyl ion production and a subsequent reduction in photocatalytic degradation efficiency. Moreover, reactive radicals and the production of degradation byproducts, which rival fluorescein for the few active sites on the NC surface, may be contributing factors to this decrease.<sup>46</sup>

**3.8.4 Effect of recycling on % degradation.** One important factor to take into account was how long the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  catalyst could withstand photocatalytic conditions. The  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  catalyst was subjected to five successive reuse cycles in order to evaluate its stability; the percentage deterioration for cycles 1 through 5 was 83.73%, 78.24%, 69.91%, 57.26%, and 44.19%, respectively (Fig. 11(d)). The catalyst was probably in its pristine state, with the best possible active sites readily available, which explained the catalyst's initial high degrading efficiency of 83.73%. The ensuing slow decrease in efficiency could be attributed to surface impurities, small structural changes, or early catalyst deactivation.

Due to the dye that is absorbed on the surface of the photocatalysts during the recycling process as well as the catalysts' mass loss, the degradation efficiency of the catalyst may somewhat decrease.<sup>47</sup> Additionally, if the catalyst surface is reused, impurities from the reaction environment may collect on it and reduce its catalytic activity. A number of reasons, including agglomeration, leaching of active species, or structural changes, may cause catalyst deactivation over the course of numerous cycles, hence decreasing the number of active sites accessible for the photocatalytic reaction.<sup>48</sup>

## 4. Conclusions

In this study,  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs have been prepared through a mechanochemical process, which is a method of treating solids and couples mechanical and chemical events at the

molecular level. Given that this procedure is more efficient, quicker, and simpler than traditional approaches, it is a better strategy for the preparation of these NCs. The physicochemical analysis shows that the crystallinity and particle size increase with increasing calcination temperature, which leads to a clear reduction in the surface area and band gap energies of the calcined analogues. As the surface area governs the photocatalytic efficiency of the nanocatalysts, and with its decreasing trend, the photocatalytic performance of the  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs decreased at elevated temperatures. The photocatalytic test was carried out under optimized conditions, and the results of the optimization process show that the calcination temperature, pH, catalyst dose, and initial concentration of dye have a great influence on the photocatalytic behavior of  $\text{Fe}_2\text{O}_3\text{-ZrO}_2$  NCs.

## Author contributions

Maqsoom Zain – formal analysis and writing – original draft; Khawaja Ansar Yasin – supervision, project administration; Sirajul Haq, Wajid Rehman and Salah Ud din – Conceptualization, methodology and project administration; Shafia Shujaat – visualization, and writing – original draft; Asad Syed and Bilal Ahamad Paray – software, data curation and funding acquisition; Jamoliddin Razzokov and Abdus Samad – validation and writing – review & editing; M. Khalid Hossain – writing – review & editing.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors would like to extend their sincere appreciation to the Researchers Supporting Project Number (RSP2024R144), King Saud University, Riyadh, Saudi Arabia.

## References

- 1 P. O. Ukaogo, U. Ewuzie and C. V. Onwuka, *Environmental Pollution: Causes, Effects, and the Remedies*, INC, 2020.
- 2 C. FN, M. MF and J. Ecosyst, *Ecography*, 2017, **07**, 5–8.
- 3 A. Tkaczyk, K. Mitrowska and A. Posyniak, *Sci. Total Environ.*, 2020, **717**, 137222.
- 4 P. J. Landrigan, R. Fuller, N. J. R. Acosta, O. Adeyi, R. Arnold, N. (Nil) Basu, A. B. Baldé, R. Bertollini, S. Bose-O'Reilly, J. I. Boufford, P. N. Breyse, T. Chiles, C. Mahidol, A. M. Coll-Seck, M. L. Cropper, J. Fobil, V. Fuster, M. Greenstone, A. Haines, D. Hanrahan, D. Hunter, M. Khare, A. Krupnick, B. Lanphear, B. Lohani, K. Martin, K. V. Mathiasen, M. A. McTeer, C. J. L. Murray, J. D. Ndahimananjara, F. Perera, J. Potočník, A. S. Preker, J. Ramesh, J. Rockström, C. Salinas, L. D. Samson, K. Sandilya, P. D. Sly, K. R. Smith, A. Steiner, R. B. Stewart, W. A. Suk, O. C. P. van Schayck, G. N. Yadama, K. Yumkella and M. Zhong, *Lancet*, 2018, **391**, 462–512.

- 5 R. N. Johnson, A. D. Fu, H. R. McDonald, J. M. Jumper, E. Ai, E. T. Cunningham and B. J. Lujan, *Fluorescein Angiography: Basic Principles and Interpretation*, Elsevier Inc., 5th edn, 2012, vol. 1.
- 6 S. Samsami, M. Mohamadi, M. H. Sarrafzadeh, E. R. Rene and M. Firoozbahr, *Process Saf. Environ. Prot.*, 2020, **143**, 138–163.
- 7 B. M. Adesanmi, H. Paul, C. Huhnke, B. M. Adesanmi, Y. Hung, H. H. Paul and C. R. Huhnke, *GSC Adv. Res. Rev.*, 2022, **10**, 126–137.
- 8 S. Ilahi, O. Manzoor, M. Mohsin and S. Ali, *Environ. Res.*, 2019, **171**, 328–340.
- 9 M. F. Warsi, N. Shaheen, M. I. Sarwar, P. O. Agboola, I. Shakir and S. Zulfiqar, *Desalin. Water Treat.*, 2021, **211**, 181–195.
- 10 P. Sangaiya and R. Jayaprakash, *J. Supercond. Novel Magn.*, 2018, **31**, 3397–3413.
- 11 O. Meza, L. A. Diaz-torres, P. Salas, E. De Rosa and D. Solis, *Mater. Sci. Eng., B*, 2010, **174**, 177–181.
- 12 O. Długosz, K. Szostak and M. Banach, *Appl. Nanosci.*, 2020, **10**, 941–954.
- 13 S. Riaz, A. Nairan and S. Naseem, *2013 World Congr. Adv. Nano, Biomech. Robot. Energy Res.*, 2013, pp. 297–305.
- 14 S. I. Siddiqui and S. A. Chaudhry, *J. Cleaner Prod.*, 2019, **223**, 849–868.
- 15 T. Friščić, C. Mottillo and H. M. Titi, *Angew. Chem.*, 2020, **132**, 1030–1041.
- 16 L. Takacs, *Prog. Mater. Sci.*, 2002, **47**, 355–414.
- 17 C. Qiu, Q. Chen, C. Chang, W. Jiang and G. Fan, *Int. J. Hydrogen Energy*, 2021, **46**, 656–665.
- 18 M. Mokhtar, S. N. Basahel and T. T. Ali, *J. Mater. Sci.*, 2013, **48**, 2705–2713.
- 19 D. J. Kim, S. H. Hahn, S. H. Oh and E. J. Kim, *Mater. Lett.*, 2002, **57**, 355–360.
- 20 C. Chan, J. Porter, Y. Li and W. Guo, *J. Am. Ceram. Soc.*, 1999, **72**, 566–572.
- 21 N. G. Menon, S. Sarma and S. Mukherji, *Appl. Nanosci.*, 2018, **8**, 915–930.
- 22 S. B. Kulkarni, U. M. Patil, R. R. Salunkhe, S. S. Joshi and C. D. Lokhande, *J. Alloys Compd.*, 2011, **509**, 3486–3492.
- 23 H. Helmiyati, Y. Budiman, G. H. Abbas, F. W. Dini and M. Khalil, *Heliyon*, 2021, **7**(3), 1–9.
- 24 H. Helmiyati, N. Fitriana, M. L. Chaerani and F. W. Dini, *Opt. Mater.*, 2022, **124**, 111982.
- 25 S. N. Basahel, M. Mokhtar, T. T. Ali and K. Narasimharao, *Catal. Today*, 2020, **348**, 166–176.
- 26 S. A. Moon, B. K. Salunke, B. Alkotaini, E. Sathiyamoorthi and B. S. Kim, *IET Nanobiotechnol.*, 2015, **9**, 220–225.
- 27 R. Karimi and M. Homayoonfal, *Polym. Adv. Technol.*, 2021, **32**, 1345–1362.
- 28 A. Ragunathan, R. Krishnan and B. Ameen, *J. Chem. Res.*, 2015, **39**, 622–626.
- 29 N. C. Horti, M. D. Kamatagi, N. R. Patil, M. N. Wari and S. R. Inamdar, *Optik*, 2018, **169**, 314–320.
- 30 S. Haq, S. Dildar, M. Ben Ali, A. Mezni, A. Hedfi, M. I. Shahzad, N. Shahzad and A. Shah, *Mater. Res. Express*, 2021, **8**, 055006.
- 31 M. A. Wahba and A. A. Badawy, *J. Sol-Gel Sci. Technol.*, 2020, **94**, 637–647.
- 32 S. Haq, F. Abbasi, M. Ben Ali, A. Hed, A. Mezni, W. Rehman, M. Waseem, A. R. Khan and H. Shaheen, *Mater. Res. Express*, 2021, **8**, 075009.
- 33 S. N. Basahel, T. T. Ali, K. Narasimharao, A. A. Bagabas and M. Mokhtar, *Mater. Res. Bull.*, 2012, **47**, 3463–3472.
- 34 Z. Vasiljević, M. P. Dojčinović, J. D. Vujančević, M. Spreitzer, J. Kovač, D. Bartolić, S. Marković, I. Janković-Čaštan, N. B. Tadić and M. V. Nikolić, *RSC Adv.*, 2021, **11**, 32358–32368.
- 35 S. Haq, W. Rehman, M. Waseem, V. Meynen, S. U. Awan, A. R. Khan, S. Hussain, Z. Ul-Abdin, S. U. Din, M. Hafeez and N. Iqbal, *J. Inorg. Organomet. Polym. Mater.*, 2021, **31**, 1312–1322.
- 36 C. Lin, H. Chen, A. Nakaruk, P. Koshy and C. C. Sorrell, *Energy Procedia*, 2013, **34**, 627–636.
- 37 S. Shoukat, W. Rehman, S. Haq, M. Waseem and A. Shah, *Mater. Res. Express*, 2019, **6**, 115052.
- 38 L. Elsellami, F. Dappozze, N. Fessi, A. Houas and C. Guillard, *Process Saf. Environ. Prot.*, 2018, **113**, 109–121.
- 39 M. Farhan Hanafi and N. Sapawe, *Mater. Today: Proc.*, 2019, **19**, 1533–1536.
- 40 N. Bibi, S. Haq, W. Rehman, M. Waseem, M. U. Rehman, A. Shah, B. Khan and P. Rasheed, *Biointerface Res. Appl. Chem.*, 2020, **10**, 5895–5900.
- 41 H. Salavati, N. Tavakkoli and M. Hosseinpoor, *Ultrason. Sonochem.*, 2012, **19**, 546–553.
- 42 D. Dodoo-Arhin, T. Asiedu, B. Agyei-Tuffour, E. Nyankson, D. Obada and J. M. Mwabora, *Mater. Today: Proc.*, 2021, **38**, 809–815.
- 43 U. G. Akpan and B. H. Hameed, *J. Hazard. Mater.*, 2009, **170**, 520–529.
- 44 D. M. Nzilu, E. S. Madivoli, D. S. Makhanu, S. I. Wanakai, G. K. Kiprono and P. G. Kareru, *Sci. Rep.*, 2023, **13**, 14030.
- 45 S. Haq, H. Afsar, I. U. Din, P. Ahmad, M. U. Khandaker, H. Osman, S. Alamri, M. I. Shahzad, N. Shahzad, W. Rehman and M. Waseem, *Catalysts*, 2021, **11**, 1–15.
- 46 S. Haq, A. Sarfraz, F. Mena, N. Shahzad, S. U. Din, H. A. Almukhlifi, S. A. Alshareef, E. M. Al Essa and M. I. Shahzad, *Molecules*, 2022, **27**, 1–19.
- 47 J. Ren, Y. Z. Wu, Y. Dai, D. W. Sha, M. Chen, J. J. Wang, J. M. Pan, H. Tang, X. N. Cheng and X. H. Yan, *Mater. Technol.*, 2017, **32**, 574–583.
- 48 S. M. Hassan, A. I. Ahmed and M. A. Mannaa, *J. Sci.: Adv. Mater. Devices*, 2019, **4**, 400–412.

