



Cite this: *RSC Adv.*, 2024, 14, 18291

# Electronic, magnetic and optical properties of the charge-disproportionated $\text{YNiO}_3$ compound calculated using the GGA+ $U$ method

Edjan Alves da Silva,<sup>a</sup> Samuel Domenech de Candido<sup>b</sup> and Miguel Abbate<sup>c</sup> \*

The electronic, magnetic and optical properties of  $\text{YNiO}_3$  were studied using the GGA+ $U$  method. This compound presents charge disproportionation with two nonequivalent Ni ion sites, namely, Ni1 with  $\text{Ni}^{(3-\delta)+}$  and Ni2 with  $\text{Ni}^{(3+\delta)+}$ . A band-structure calculation was performed for the observed  $P2_1/n$  monoclinic phase; a  $2 \times 1 \times 2$  supercell with 80 atoms was used to reproduce the observed magnetic ordering. The density of states (DOS) calculation clearly showed the bonding differences between the Ni1 and the Ni2 sites; the Ni1 site has a more ionic character, whereas the Ni2 site presents a more covalent character. The band-structure results correspond to an insulator phase with a band gap of about 0.5 eV. The magnetic ordering is antiferromagnetic and the magnetic moments are about  $1.24 \mu_B$  for Ni1 and around  $0.00 \mu_B$  for Ni2. The calculated magnetic ordering results are in good agreement with neutron diffraction results. The calculated optical conductivity is similar to previous experimental data.

Received 14th March 2024

Accepted 9th May 2024

DOI: 10.1039/d4ra01908c

rsc.li/rsc-advances

## 1 Introduction

The rare-earth nickelates present a broad range of very interesting physical properties. These include, for instance, a metal-insulator transition, diverse structural and magnetic transitions, as well as charge disproportionation.<sup>1–3</sup> This variety is partly related to the interplay between the electronic, magnetic and structural degrees of freedom in these compounds. In addition, the highly correlated electronic structures of the rare-earth nickelates adds to the complexity of their physical properties. To this end, the study of the relationship between their electronic structures and their physical properties is highly desirable. The aim of this work is to study the electronic structure of the  $\text{YNiO}_3$  compound using the GGA+ $U$  method. The theoretical results are compared to the experimental electronic, magnetic and optical properties of this compound.

The  $\text{YNiO}_3$  oxide presents a metal-insulator phase transition around 582 K, which also involves structural and magnetic ordering changes.<sup>4–6</sup> In the low-temperature phase, the  $\text{YNiO}_3$  compound is an antiferromagnetic insulator (AFM-I); the electrical conductivity and the magnetic susceptibility results reflect electron-correlation effects.<sup>7</sup> The crystal structure in this phase is monoclinic with space group  $P2_1/n$ ;<sup>5</sup> the schematic representation of the crystalline structure of  $\text{YNiO}_3$  is shown in Fig. 1.

The AFM-I phase is characterized by a distortion of the  $\text{NiO}_6$  octahedra, which gives rise to two nonequivalent Ni sites: Ni1 sites with larger Ni–O bonds and Ni2 sites with shorter Ni–O bonds. This difference was attributed to charge disproportionation with charge ordering leading to  $\text{Ni}^{(3-\delta)+}$  and  $\text{Ni}^{(3+\delta)+}$  ionic states.<sup>5,8,9</sup> Hereafter, all the calculations and comparisons will be only concerned with the low-temperature AFM-I phase.

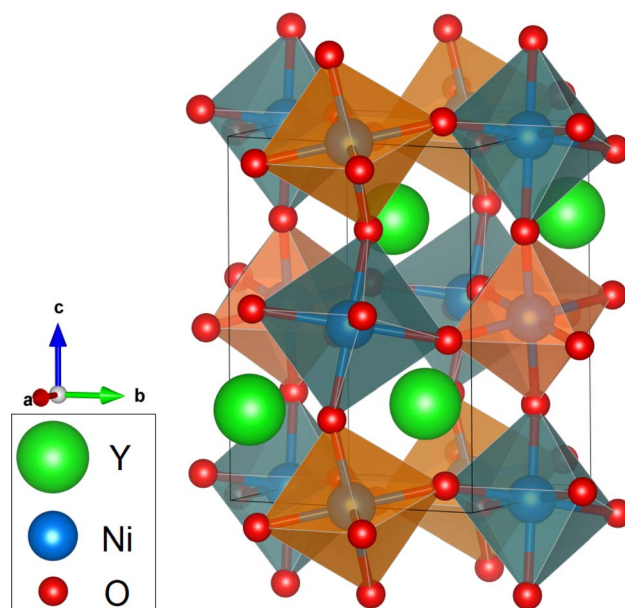


Fig. 1 Schematic VESTA representation of the crystalline structure of  $\text{YNiO}_3$  with shorter and larger Ni–O bonds indicated by orange and dark green octahedra, respectively.<sup>10</sup>

<sup>a</sup>Instituto de Física, Universidade Federal do Rio Grande do Sul, Av. Bento Gonçalves 9500, 91501-970 Porto Alegre, RS, Brazil

<sup>b</sup>Instituto Federal de Educação, Ciência e Tecnologia, Campus Campo Grande, Rua Taquari 831, 79100-510 Campo Grande, MS, Brazil

<sup>c</sup>Departamento de Física, Universidade Federal do Paraná, Caixa Postal 19044, 81531-990 Curitiba, PR, Brazil. E-mail: miguel@fisica.ufpr.br



The electronic structure of  $\text{YNiO}_3$  was studied using soft X-ray resonant magnetic powder diffraction,<sup>11</sup> X-ray absorption spectroscopy (XAS)<sup>12</sup> and optical absorption.<sup>7</sup> The magnetic and structural properties of this oxide were investigated using synchrotron radiation diffraction,<sup>13</sup> neutron and synchrotron radiation diffraction,<sup>5,14</sup> and electron spin resonance (ESR) and susceptibility.<sup>15,16</sup> The electronic structure of  $\text{YNiO}_3$  was calculated using several band-structure methods, including: LDA,<sup>17</sup> LDA+ $U$ ,<sup>18</sup> LSDA+ $U$ ,<sup>19</sup> GGA+ $U$ ,<sup>20</sup> and the Hartree–Fock approach.<sup>21</sup> Recent results include studies of the optical spectra,<sup>22</sup> and additional band-structure calculations.<sup>23</sup>

Considerable insight on the properties of  $\text{YNiO}_3$  was obtained from the above band structure and by modeling Hamiltonian models. But a comprehensive comparison of the calculations and the experimental results is still lacking and would be desirable. This work presents the electronic structure of  $\text{YNiO}_3$  calculated by the GGA+ $U$  method. The results are in reasonable agreement with the available experimental data, and help to explain the electronic, magnetic and optical properties of this compound. The results are also interpreted in terms of the configuration interaction (CI) scheme of highly correlated  $pd$  models. Finally, the interplay between the structural, charge and spin degrees of freedom is discussed using the CI language.

## 2 Calculation details

Calculations for the AFM-I phase at room temperature were performed using the WIEN2k package,<sup>24</sup> mainly because most of the available experimental results were obtained under this condition. This program is based in the full-potential linearized augmented plane wave (LAPW) method. The GGA exchange–correlation potential was obtained using the PBEsol scheme,<sup>25</sup> and the on-site Hubbard parameter  $U$  was set to 1.5 eV. This value was chosen because it yields the best overall agreement with the experimental results. We also note that this value is in line with those previously used for this compound.<sup>19,20,26</sup>

The spin–orbit splitting of the Ni 3d orbitals was included to determine the magnetic ordering. The spin–orbit coupling determines the orientation of the spin, so it is crucial for the calculation of the magnetic ordering. But the value of the coupling constant (about 0.04 eV for Ni) is relatively small, and does not affect much the electronic structure or the physical properties. The calculation of the magnetic ordering was unconstrained, and all the magnetic moments were allowed to rotate, until the criterion of total energy convergence was achieved.

The crystal structure was monoclinic and the space group was  $P2_1/n$ . The lattice parameters are listed in Table 1; the

**Table 1** Comparison between the experimental (taken from Alonso *et al.*<sup>5</sup>) and calculated lattice parameters in  $\text{YNiO}_3$  (all values in Å)

	Experimental	Calculated
<i>a</i>	5.17932	5.13105
<i>b</i>	5.51529	5.46389
<i>c</i>	7.41656	7.34744

**Table 2** Comparison between the experimental (taken from Alonso *et al.*<sup>5</sup>) and calculated relative atomic positions of the different nonequivalent atoms in  $\text{YNiO}_3$  referred to the monoclinic axes

Atom	Experimental			Calculated		
	<i>x</i>	<i>y</i>	<i>z</i>	<i>x</i>	<i>y</i>	<i>z</i>
Y	0.9816	0.0729	0.2502	0.9796	0.0735	0.2505
Ni1	0.5000	0.5000	0.0000	0.5000	0.5000	0.0000
Ni2	0.5000	0.0000	0.5000	0.5000	0.0000	0.5000
O1	0.0998	0.4705	0.2457	0.0977	0.4729	0.2504
O2	0.6973	0.3080	0.0467	0.6942	0.3011	0.0472
O3	0.1882	0.2038	0.9465	0.1952	0.2002	0.9501

experimental values were retrieved from Alonso *et al.*,<sup>5</sup> and the theoretical parameters were obtained by a total energy calculation. The cell volume was reduced about 2.8% from 211.85 Å<sup>3</sup> (experimental) to 205.98 Å<sup>3</sup> (calculated).

The nonequivalent atomic positions are given in Table 2; the experimental results were retrieved from Alonso *et al.*<sup>5</sup> and the theoretical positions were obtained by an atomic force calculation (the convergence criteria for the force was 1 mRy Bohr<sup>−1</sup>). The experimental and calculated average Ni–O bond lengths are in excellent agreement: 1.994 Å (exper.) vs. 1.993 Å (theor.) at the Ni1 site, and 1.923 Å (exper.) vs. 1.922 Å (theor.) at the Ni2 site.

The maximum wavevector of the plane-wave basis set was given by  $R_{\text{MT-KMAX}} = 7.0$ , the integration was performed using 500 *k*-points in the irreducible part of the Brillouin zone, the energy convergence criteria was set to 10<sup>−5</sup> Ry, and the 2 × 1 × 2 super-cell was formed by 80 atoms (please note that the monoclinic cell contains four chemical formulas with 20 atoms).

## 3 Results and discussion

### 3.1 Band structure and band gap

Fig. 2 shows the projected density of states (DOS) of the  $\text{YNiO}_3$  compound, separated into the Ni1 (upper panel) and the Ni2 (lower panel) nonequivalent sites.<sup>5,13,14</sup> These results correspond to a stable and fully converged antiferromagnetic insulating (AFM-I) phase. The DOS for each nonequivalent site are split into the majority (above) and minority (below) spin bands. The Fermi level ( $E_F = 0$  eV) separates the occupied states in the valence band and the unoccupied states in the conduction band. The Ni 3d and the O 2p bands present a strong covalent mixing from −7 to +3 eV, whereas the Y 4d bands exhibit a more ionic character from +5 to +9 eV.

The Ni 3d bands of the Ni1 and Ni2 nonequivalent sites show clear differences; see the inset in Fig. 2. In particular, the Ni 3d states contribute more to the bonding part of the Ni 3d–O 2p bands, from −7 to −4 eV, in the Ni2 site (lower panel). This means that, as expected, the Ni1 site is slightly more ionic, whereas the Ni2 site is more covalent; see below.

The calculated DOS presented here are in good agreement with previous results in the literature.<sup>8,18,19,27</sup> The calculated band gap between the valence and the conduction band, about 0.5 eV, agrees with the optical band gap measured by Arima and



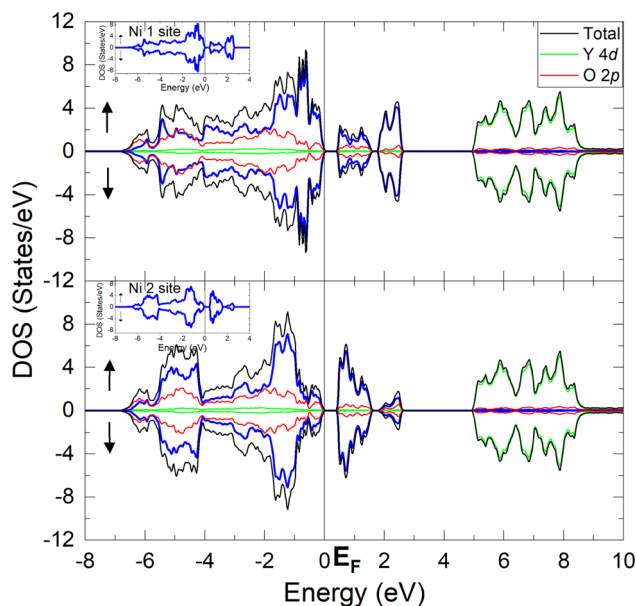


Fig. 2 Projected density of states (DOS) of the  $\text{YNiO}_3$  compound separated into the Ni1 (upper panel) and the Ni2 (lower panel) nonequivalent sites. The DOS in each panel are split in the majority (above) and minority (below) spin bands. The Fermi level separates the occupied states in the valence band from the unoccupied states in the conduction band.

Tokura,<sup>28</sup> around 0.3 eV, although the present calculation slightly overestimated the value of the experimental band gap. We note that the value of the band gap found here is similar to those reported in recent studies.<sup>19,20,26</sup>

### 3.2 Magnetic moments and magnetic ordering

The calculated magnetic moments for the Ni ions at the Ni1 sites are about  $1.24 \mu_B$  and for those at the Ni2 sites they are around  $0.00 \mu_B$ . A comparison with the experimental values obtained by Alonso *et al.*<sup>5</sup> using neutron diffraction is shown in Table 3. The results at the Ni1 site, 1.4 (exper.) vs.  $1.24 \mu_B$  (theor.), are in reasonable agreement, but the values at the Ni2 site, 0.7 (exper.) vs.  $0.00 \mu_B$  (theor.), are in clear disagreement. It is worth noting that recent calculations by other groups also yield a negligible magnetic moment at the Ni2 site.<sup>29,30</sup> Additional work, both theoretical and experimental, would be desirable to reconcile this discrepancy.

Fig. 3 compares the antiferromagnetic ordering of the  $\text{YNiO}_3$  oxide at room temperature obtained by Alonso *et al.*<sup>5</sup> (top) with the result of the present calculation produced with the VESTA program<sup>10</sup> (bottom). The small arrows at the Ni2 sites represent

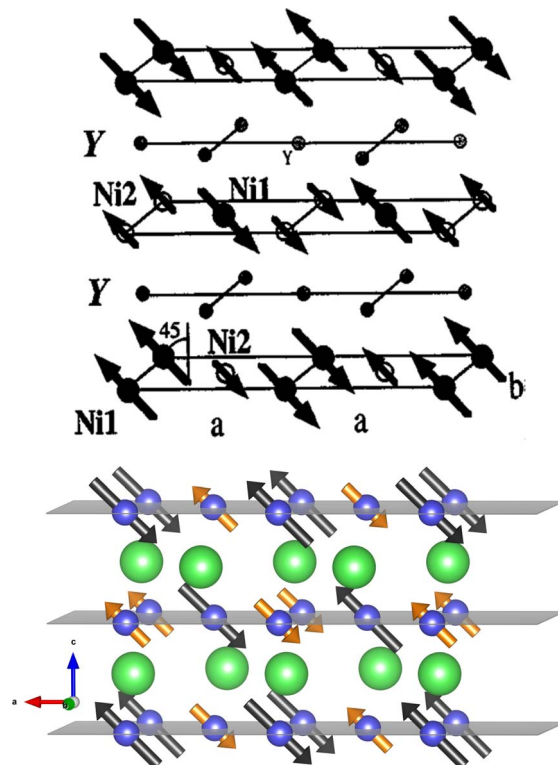


Fig. 3 Antiferromagnetic ordering of the  $\text{YNiO}_3$  perovskite at room temperature obtained by Alonso *et al.*<sup>5</sup> (top) compared to the present calculated result produced with the VESTA program<sup>10</sup> (bottom).

the orientation of the residual  $0.0083 \mu_B$  magnetic moments given by the calculation. Despite the discrepancy in the value of the magnetic moment at the Ni2 site, there is an overall good agreement in the relative orientation of these moments. Please note that the use of the  $2 \times 1 \times 2$  supercell structure with 80 atoms and the inclusion of the spin-orbit coupling of the Ni 3d orbitals were crucial to reproduce the experimental magnetic ordering.

### 3.3 Optical conductivity

Fig. 4 compares the optical conductivity obtained by Arima and Tokura<sup>28</sup> at room temperature and the calculated value of  $\sigma(\omega)$  for the  $\text{YNiO}_3$  compound. The experimental optical conductivity of the  $\text{YNiO}_3$  compound presents a narrow band gap of about 0.3 eV,<sup>28</sup> whereas the calculated result yields a slightly larger band gap of around 0.5 eV. Both the experimental and calculated optical conductivity values reflect the insulating character of the  $\text{YNiO}_3$  compound at room temperature. Please note the absence of the Drude peak in the spectra, which is a characteristic of a metallic phase. The increase of  $\sigma(\omega)$  at higher energies, above 5 eV, is attributed to the onset of interband O 2p to Ni 3d transitions.

### 3.4 Configuration interaction description

The electronic state of the divalent  $\text{Ni}^{2+}$  ions in NiO oxide is mainly represented by an ionic  $3d^8$  configuration.<sup>31,32</sup> However, the trivalent  $\text{Ni}^{3+}$  ions in the  $\text{LaNiO}_3$  compound are not

Table 3 Comparison between the calculated and experimental values of the magnetic moments in the  $\text{YNiO}_3$  compound

Magnetic moments ( $\mu_B$ )		
Ni sites	Experimental <sup>5</sup>	This work
Ni1	1.4	1.24
Ni2	0.7	0.00



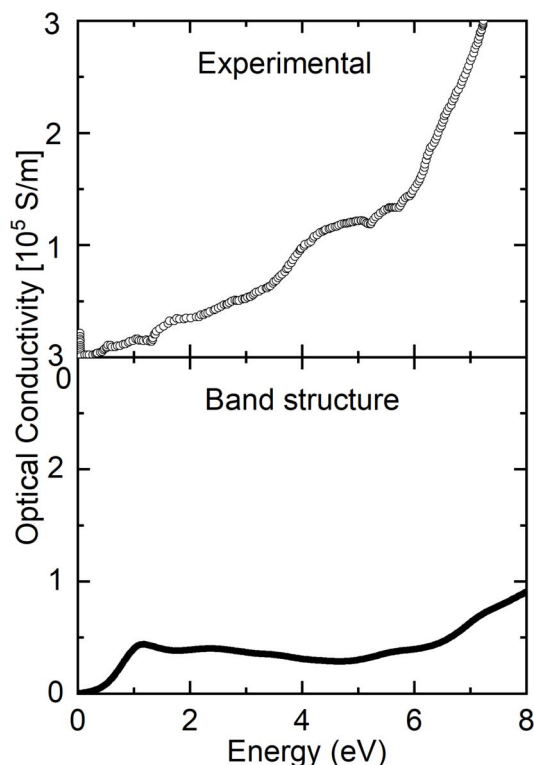


Fig. 4 Experimental optical conductivity extracted from Arima and Tokura (ref. 28) compared to the calculation of the  $\text{YNiO}_3$  compound.

dominated by the ionic  $3d^7$  configuration, but rather by the covalent  $3d^8\bar{L}$  configuration, where  $\bar{L}$  corresponds to a ligand (O 2p) hole.<sup>33,34</sup> This means that the third hole does not go to the Ni 3d states, as expected in an ionic approximation, but rather to the O 2p orbitals, which reflects the larger covalent contribution to the bonding in this material. In terms of the highly correlated pd models used to study these materials, this corresponds to the so-called negative charge-transfer regime.<sup>35</sup>

In the  $\text{LaNiO}_3$  compound, the oxygen octahedra are similar and the  $3d^8\bar{L}$  configuration is uniformly spread throughout the crystal. In the  $\text{YNiO}_3$  oxide, there is a structural distortion in the octahedra, which produces two nonequivalent Ni sites. The ensuing charge disproportionation is the result of a charge redistribution between the  $\text{Ni}^{(3-\delta)+}$  site and the  $\text{Ni}^{(3+\delta)+}$  site. The charge redistribution does not involve the strongly bound Ni 3d states, but rather the loosely bound hole in the O 2p orbitals. In particular, the Ni1 site (with a larger octahedron and weaker Ni 3d–O 2p interactions) is better described by a  $3d^8$  configuration, whereas the Ni2 sites (with a smaller octahedron and stronger Ni 3d–O 2p interactions) is mainly given by a  $3d^8\bar{L}^2$  configuration.<sup>26,36</sup>

The DFT band-structure method is an independent particle approach, but it is a first-principles method and takes into account translational invariance. On the other hand, the highly correlated pd models consider many-body effects, but they are based on adjustable parameters and are often applied to small systems. This discussion illustrates that they have complementary advantages and limitations. In the present work, the first method is mostly used because it provides straightforward results that can be compared with the experimental results.

Table 4 Relationships between the structural (Ni–O bonds), charge (Configuration) and spin (Spin) degrees of freedom for the Ni1 and the Ni2 nonequivalent sites of the  $\text{YNiO}_3$  compound

Degrees of freedom			
Ni sites	Ni–O bonds	Configuration	Spin
Ni1	Larger	$3d^8$	$S = 1$
Ni2	Shorter	$3d^8\bar{L}^2$	$S = 0$

### 3.5 Structural, charge and spin interactions

The  $\text{YNiO}_3$  material presents a clear interaction between the structural, charge and spin degrees of freedom. The structural distortion produces a Ni1 site at the larger octahedra with larger Ni–O bonds, and a Ni2 site at the smaller octahedra with smaller Ni–O bonds. This results in charge disproportionation with a more ionic Ni1 site with a  $3d^8$  configuration, and a more covalent Ni2 site with a  $3d^8\bar{L}^2$  configuration. The spin at the Ni1 site, with the  $3d^8$  configuration, corresponds to a  $S = 1$  high-spin state, whereas the spin at the Ni2 site, with the  $3d^8\bar{L}^2$  configuration, is given by a  $S = 0$  low-spin state.<sup>26,36</sup> In turn, this helps to explain the calculated  $1.24 \mu_B$  magnetic moment at the Ni1 site, and the  $0.00 \mu_B$  magnetic moment at the Ni2 site. For an easier reference, the relationships between the Ni–O bonds, the electronic configuration and the spin state for each nonequivalent site are listed in Table 4.

### 3.6 Charge disproportionation

Alonso *et al.*<sup>5</sup> used the phenomenological Brown bond-valence model to estimate the Ni valence; they obtained two different values of the delta parameter: around  $\delta = 0.38$  at the Ni1 site and about  $\delta' = 0.17$  at the Ni2 site. Later on, in the analysis of the different models used to interpret the neutron diffraction results, they concluded that  $\delta = \delta' = 0.35$ .<sup>5</sup> The present band-structure calculation gives a smaller estimate of around  $\delta = 0.05$ . But, as discussed above, a double-cluster method predicts small changes at the Ni sites and larger changes at the ligand (O 2p) band<sup>36</sup> (this is reasonable because O 2p–O 2p charge fluctuations cost less energy than Ni 3d–Ni 3d charge variations). Using this method, we estimate that the ligand contribution is about  $\delta = 0.3$ .<sup>36</sup> The combination of  $\delta = 0.05$  from the band structure and  $\delta = 0.3$  from the double-cluster calculations is similar to the value  $\delta = 0.35$  obtained from neutron diffraction data by Alonso *et al.*<sup>5</sup>

## 4 Summary and conclusions

In summary, we studied the electronic structure of the  $\text{YNiO}_3$  compound using the GGA+*U* method with  $U = 1.5$  eV. In particular, we investigated the charge disproportionation between the nonequivalent Ni1 and Ni2 sites. The stable phase at room temperature is an antiferromagnetic insulating (AFM-I) state. The density of states (DOS) at the nonequivalent Ni1 and the Ni2 sites present clear differences; in particular, the Ni 3d–O 2p bonds are more ionic at the Ni1 site and more covalent at the Ni2 site. The calculated band-gap energy is in reasonable



agreement with the experimental value. The calculated magnetic moment at the Ni1 site is in agreement with the experimental value, but there is a discrepancy in the calculated moment at the Ni2 site. Despite this disagreement, the calculated magnetic ordering agrees with the neutron diffraction results. We note that the use of a  $2 \times 1 \times 2$  supercell structure with 80 atoms and the inclusion of the Ni 3d spin-orbit coupling were crucial to obtain the orientation of the magnetic moments. The calculated optical conductivity is in reasonable agreement with the experimental result. The results are also interpreted in terms of the configuration interaction (CI) scheme of highly correlated pd models. Finally, the interactions between the structural, charge and spin degrees of freedom in  $\text{YNiO}_3$  are also discussed.

## Conflicts of interest

There are no conflicts of interest to declare.

## Acknowledgements

This work was partially supported by the Brazilian funding agencies Conselho Nacional de Desenvolvimento Científico e Tecnológico (CNPq) and Coordenação de Aperfeiçoamento de Pessoal de Nível Superior (CAPES).

## References

- J. B. Torrance, P. Lacorre, A. I. Nazzari, E. J. Ansaldo and C. Niedermayer, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **45**, 8209–8212.
- G. Catalan, *Phase Transitions*, 2008, **81**, 729–749.
- M. L. Medarde, *J. Phys.: Condens. Matter*, 1997, **9**, 1679–1707.
- J. L. García-Muñoz, J. Rodríguez-Carvajal, P. Lacorre and J. B. Torrance, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **46**, 4414.
- J. A. Alonso, J. L. García-Muñoz, M. T. Fernández-Díaz, M. A. G. Aranda, M. J. Martínez-Lope and M. T. Casais, *Phys. Rev. Lett.*, 1999, **82**, 3871–3874.
- J.-S. Zhou and J. B. Goodenough, *Phys. Rev. Lett.*, 2005, **94**, 065501.
- T. Arima, Y. Tokura and J. B. Torrance, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **48**, 17006–17009.
- I. I. Mazin, D. I. Khomskii, R. Lengsdorf, J. A. Alonso, W. G. Marshall, R. M. Ibberson, A. Podlesnyak, M. J. Martínez-Lope and M. M. Abd-Elmeguid, *Phys. Rev. Lett.*, 2007, **98**, 176406.
- U. Staub, G. I. Meijer, F. Fauth, R. Allenspach, J. G. Bednorz, J. Karpinski, S. M. Kazakov, L. Paolasini and F. d'Acapito, *Phys. Rev. Lett.*, 2002, **88**, 126402.
- K. Momma and F. Izumi, *J. Appl. Crystallogr.*, 2011, **44**, 1272–1276.
- Y. Bodenthin, U. Staub, C. Piamonteze, M. García-Fernández, M. Martínez-Lope and J. Alonso, *J. Phys.: Condens. Matter*, 2011, **23**, 036002.
- A. Y. Ramos, C. Piamonteze, H. C. Tolentino, N. M. Souza-Neto, O. Bunau, Y. Joly, S. Grenier, J.-P. Itié, N. E. Massa, J. A. Alonso, *et al.*, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 045102.
- J. A. Alonso, M. J. Martínez-Lope, M. T. Casais, M. A. G. Aranda and M. T. Fernández-Díaz, *J. Am. Chem. Soc.*, 1999, **121**, 4754–4762.
- J. A. Alonso, M. J. Martínez-Lope, M. T. Casais, J. L. García-Muñoz and M. T. Fernández-Díaz, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **61**, 1756.
- M. Causa, R. Sánchez, M. Tovar, J. Alonso and M. Martínez-Lope, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **68**, 024429.
- R. Sánchez, M. Causa, M. Tovar, J. Alonso and M. Martínez-Lope, *J. Magn. Magn. Mater.*, 2004, **272–276**, 390–391.
- X. Xu, X. Meng, C. Wang, F. Wu and G. Chen, *J. Phys. Chem. B*, 2004, **108**, 1165–1167.
- Y. Quan, V. Pardo and W. E. Pickett, *Phys. Rev. Lett.*, 2012, **109**, 216401.
- S. Yamamoto and T. Fujiwara, *J. Phys. Chem. Solids*, 2002, **63**, 1347–1351.
- A. Mercy, J. Bieder, J. Íñiguez and P. Ghosez, *Nat. Commun.*, 2017, **8**, 1677.
- T. Mizokawa, D. I. Khomskii and G. A. Sawatzky, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **61**, 11263–11266.
- J. Bieder, A. Mercy, W.-Y. Tong and P. Ghosez, *Phys. Rev. B*, 2020, **102**, 081111.
- L. Binci, M. Kotiuga, I. Timrov and N. Marzari, *Phys. Rev. Res.*, 2023, **5**, 033146.
- P. Blaha, K. Schwarz, G. Madsen, D. Kvasnicka and J. Luitz, *WIEN2k: An Augmented Plane Wave Plus Local Orbitals Program for Calculating Crystal Properties*, Technische Universität Wien, Wien, 2001, vol. 28.
- J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- J. Varignon, M. N. Grisolia, J. Íñiguez, A. Barthélémy and M. Bibes, *npj Quantum Mater.*, 2017, **2**, 21.
- A. Hampel and C. Ederer, *Phys. Rev. B*, 2017, **96**, 165130.
- T. h. Arima and Y. Tokura, *J. Phys. Soc. Jpn.*, 1995, **64**, 2488–2501.
- J. Du, S.-L. Shang, Y. Wang, A. Zhang, S. Xiong, F. Liu and Z.-K. Liu, *AIP Adv.*, 2021, **11**, 015028.
- J. Du, O. Malyi, S.-L. Shang, Y. Wang, X.-G. Zhao, F. Liu, A. Zunger and Z.-K. Liu, *Mater. Today Phys.*, 2022, **27**, 100805.
- G. A. Sawatzky and J. W. Allen, *Phys. Rev. Lett.*, 1984, **53**, 2339–2342.
- A. Fujimori and F. Minami, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1984, **30**, 957–971.
- M. Abbate, G. Zampieri, F. Prado, A. Caneiro, J. M. Gonzalez-Calbet and M. Vallet-Regi, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **65**, 155101.
- E. Alves, H. Martins, S. Domenech and M. Abbate, *Phys. Lett. A*, 2019, **383**, 2952–2956.
- P. Kuiper, G. Kruizinga, J. Ghijsen, G. A. Sawatzky and H. Verweij, *Phys. Rev. Lett.*, 1989, **62**, 221–224.
- R. J. Green, M. W. Haverkort and G. A. Sawatzky, *Phys. Rev. B*, 2016, **94**, 195127.

