


 Cite this: *RSC Adv.*, 2024, **14**, 13374

# NiO<sub>x</sub>/PANI nanocomposite doped carbon paste as electrode for long-term stable and highly efficient perovskite solar cells†

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Carbon-based perovskite solar cells (PSCs) have emerged as a hopeful alternative in the realm of photovoltaics. They are considered promising due to their affordability, remarkable durability in humid environments, and impressive electrical conductivity. One approach to address the cost issue is to use affordable counter electrodes in PSCs that do not require organic hole transport materials (HTMs). This study utilized an innovative and economical method to create NiO<sub>x</sub>/PANI nanocomposites. Later, these nanoparticles were integrated into a carbon paste to act as an HTM. This incorporation is intended to optimize charge extraction, improve interfacial contact, align energy levels, reduce energy loss, minimize charge recombination, and protect the perovskite (FAPbI<sub>3</sub>) surface from degradation. The optoelectronic properties of these devices were investigated, and all cells showed improved efficiency compared to control cells. The NiO<sub>x</sub>/PANI doped carbon (NiO<sub>x</sub>/PANI+CE) exhibited excellent performance due to strong hole conductivity, well-aligned energy levels, and the formation of stepwise band alignment at the perovskite interface.

Received 19th February 2024

Accepted 13th April 2024

DOI: 10.1039/d4ra01287a

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## 1. Introduction

Photovoltaic devices are recognized as an innovative energy source. Current solar cell generations that utilize solution-based methods, such as organic, dye-sensitized, quantum dot, and perovskite-based technologies,<sup>1-4</sup> have made significant advances in recent years compared to traditional silicon- and thin-film-based solar

cells.<sup>5,6</sup> Metal halide perovskite solar cells have been extensively studied, have shown considerable performance improvements, and are the subject of further commercial investigation.<sup>7,8</sup>

Perovskite solar cells are recognized for their usefulness and efficiency.<sup>9</sup> Organic components such as spiro-OMeTAD, PTAA, and PCBM have been utilized for electron and hole transport in PSCs.<sup>10</sup> However, these materials have significant limitations, including complex and expensive synthesis, possible low mobility, poor crystallinity, and possible degradation due to environmental factors. Therefore, there is a great desire to replace these with nanocomposites.<sup>11,12</sup>

Although major efforts have been made to improve the efficiency of PSCs, concerns remain about their cost and the need for high-temperature processing, which may hinder their widespread use. The high cost is due to the complex synthesis process and strict purity requirements of organic hole transport materials (OHTMs) such as spiro-OMeTAD and PTAA, as well as the use of noble metal electrodes such as gold and silver.<sup>13</sup> Additionally, high-temperature annealing steps increase production costs and energy consumption, especially in mesoscopic devices. The use of OHTMs-free PSCs and low-cost counter electrodes containing nickel and carbon is one tactic to overcome these problems.<sup>14-16</sup> This approach has the potential to reduce costs and simplify the manufacturing process. Among the various inorganic materials used for charge transport, NiO<sub>x</sub> stands out as a very affordable and abundant p-type semiconductor. Its properties include a wide bandgap, high

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† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4ra01287a>



hole mobility, strong thermal and chemical stability, and good energy compatibility with carbon materials and perovskites.<sup>17,18</sup>

Moreover, the incorporation of NiO<sub>x</sub> NPs into carbon electrodes, such as single-walled carbon nanotubes or graphite/carbon black, effectively promoted the extraction of holes between the perovskite layer and the back electrode.<sup>19</sup> Additionally, the use of two-dimensional carbon materials in PSC structures may lead to unwanted charge recombination through poor layer-to-layer contact.<sup>20</sup> To maximize the benefits, one effective approach is to react carbon nanomaterials with a polymer material, forming composites.<sup>21</sup> These composites exhibit a synergistic effect between the nanofillers and polymer, resulting in boosted properties under various conditions.

Currently, the primary focus on regulating the performance of carbon electrodes for PSCs has centered around the selection of graphite and carbon black. Limited research has been conducted to explore the enhancement of their photovoltaic properties through surface treatment. Various tactics including doping p-type nanoparticles (NPs), bilayer engineering, and material processing might increase efficiency.<sup>22–25</sup> Additionally, a key factor in achieving high conductivity and sufficient porosity for precursor infiltration is proper sintering.<sup>26–28</sup> The cathode layer's conductivity and hole extraction are successfully increased by the p-type nanostructured additives (such as NiO, CuS, CuSCN, and CuI) with comparatively high hole mobility.<sup>25,29–31</sup> The introduction of surface B doping and bulk P doping may endow high charge carrier mobility. Recent studies have shown that the role

of P atoms could enhance n-type behavior and B atoms could promote hole extraction.<sup>27–29</sup> The main function of doping is to lower or raise the Fermi level of carbon *via* the addition of donors or acceptors, respectively.<sup>25</sup> In addition, the heteroatom doping-induced charge redistribution, regardless of a higher (such as N) or lower (such as B, P) electronegativity than that of carbon, could break electroneutrality and enhance the conductivity of carbon.<sup>22,26</sup> In the perovskite layer, carbon-polymer nanocomposites can play various roles. As an example, Rajamanickam *et al.* reported depositing a graphene–PANI composite on top of the CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> layer to protect it from degradation.<sup>32</sup>

In this study, we introduce a novel method for constructing a carbon electrode by integrating a NiO<sub>x</sub>/PANI nanocomposite onto the carbon surface using *in situ* polymerization. This structural modification enhances conductivity and improves the contact characteristics with the perovskite, resulting in a carbon electrode system with a strong ability to collect holes. The outcomes demonstrate significant advancements in photovoltaic parameters, leading to an increase in the power conversion efficiency (PCE) from 14.46% (control devices) to 18.30%. Notably, the modified carbon electrode also significantly enhances the PSCs' stability. These findings highlight a promising method for enhancing the optoelectrical properties of carbon electrodes in PSCs.

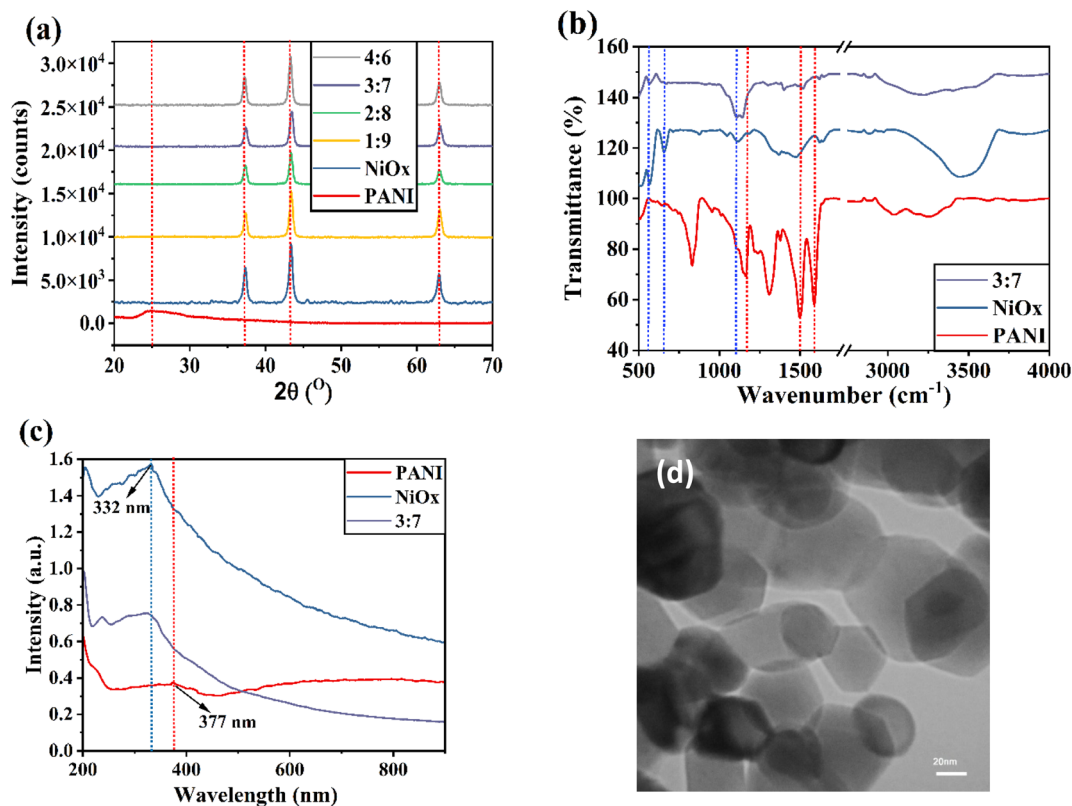


Fig. 1 (a) XRD patterns of different materials. (b) FTIR spectra and (c) UV-Vis spectra of PANI, NiO<sub>x</sub>, and NiO<sub>x</sub>/PANI (3 : 7) materials. (d) TEM image of NiO<sub>x</sub> nanoparticles.

## 2. Results

Fig. 1a depicts the XRD patterns of PANI, NiO<sub>x</sub> nanoparticles, and NiO<sub>x</sub>/PANI (3 : 7 mass/mass). In the range of  $2\theta$  from 20 to 70°, PANI exhibits a broad peak at 25.2°, corresponding to the (322) lattice plane.<sup>22</sup> NiO<sub>x</sub> nanoparticles with a cubic structure show diffraction peaks at 37.40° (111), 43.44° (200), and 63.01° (220).<sup>23</sup> In Fig. S1 and S2,<sup>†</sup> the XRD patterns of the NiO<sub>x</sub>/PANI [1 : 9 (mass/mass)] suggest that the intensity of the peaks of nickel nanoparticles is intense, while the peaks of PANI are seen with very low intensity in the composite. The characteristic peaks of NiO<sub>x</sub> shift to higher angles in the XRD pattern of the NiO<sub>x</sub>/PANI (3 : 7) due to the interaction of PANI chains and NiO<sub>x</sub> nanoparticles. Fig. 1b depicts the FTIR spectra of PANI, NiO<sub>x</sub>, and NiO<sub>x</sub>/PANI (3 : 7) materials.

In the FTIR spectra of NiO<sub>x</sub>, the main characteristic peaks were found at 564, 656, and 1110 cm<sup>-1</sup>, assigned to the vibration mode of the Ni–O band.<sup>33</sup> The strong peaks observed in the range of 1100–1600 cm<sup>-1</sup> in the FTIR spectra of PANI can be correlated to the quinonoid and benzenoid vibrational bands. Specifically, the main peaks at 1584 cm<sup>-1</sup> and 1497 cm<sup>-1</sup> correspond to the C=N and C=C stretching vibration modes, respectively. Additionally, the peaks around 1170 cm<sup>-1</sup> are

linked to the C–N benzenoid ring stretching mode.<sup>34–36</sup> When comparing the FTIR spectra of NiO<sub>x</sub>/PANI materials to those of PANI and NiO<sub>x</sub> individually, some noticeable changes can be observed. Firstly, there is a small shift in the wavenumber of the peaks, indicating a modification in the molecular structure. Secondly, the intensity of the peaks also changes, suggesting alterations in the bonding environment. For the nanocomposites NiO<sub>x</sub>/PANI, the peaks detected at 1620 cm<sup>-1</sup> and 1520 cm<sup>-1</sup> can be attributed to the C=N and C=C vibration modes of for various weight ratios of the nanocomposites, indicating the presence of both quinonoid and benzenoid units. Furthermore, the peak around 1147 cm<sup>-1</sup> is associated with the C–N stretching mode of the benzenoid ring.<sup>37,38</sup> An interesting observation is the appearance of bands at 573 and 1106 cm<sup>-1</sup>, which can be attributed to the presence of NiO<sub>x</sub> in the nanocomposite. This indicates a strong interaction between the PANI molecules and the NiO<sub>x</sub> NPs. Hence, it can be concluded that there are significant reactions between NiO<sub>x</sub> and PANI materials, leading to the formation of NiO<sub>x</sub>/PANI within the PANI matrix. Fig. 1c shows the absorbance spectra of PANI, NiO<sub>x</sub>, and NiO<sub>x</sub>/PANI. As shown, the PANI sample has a characteristic peak at 377 nm and the NiO<sub>x</sub> sample has a peak at 332 nm. In the absorbance spectra of NiO<sub>x</sub>/PANI material, a peak around

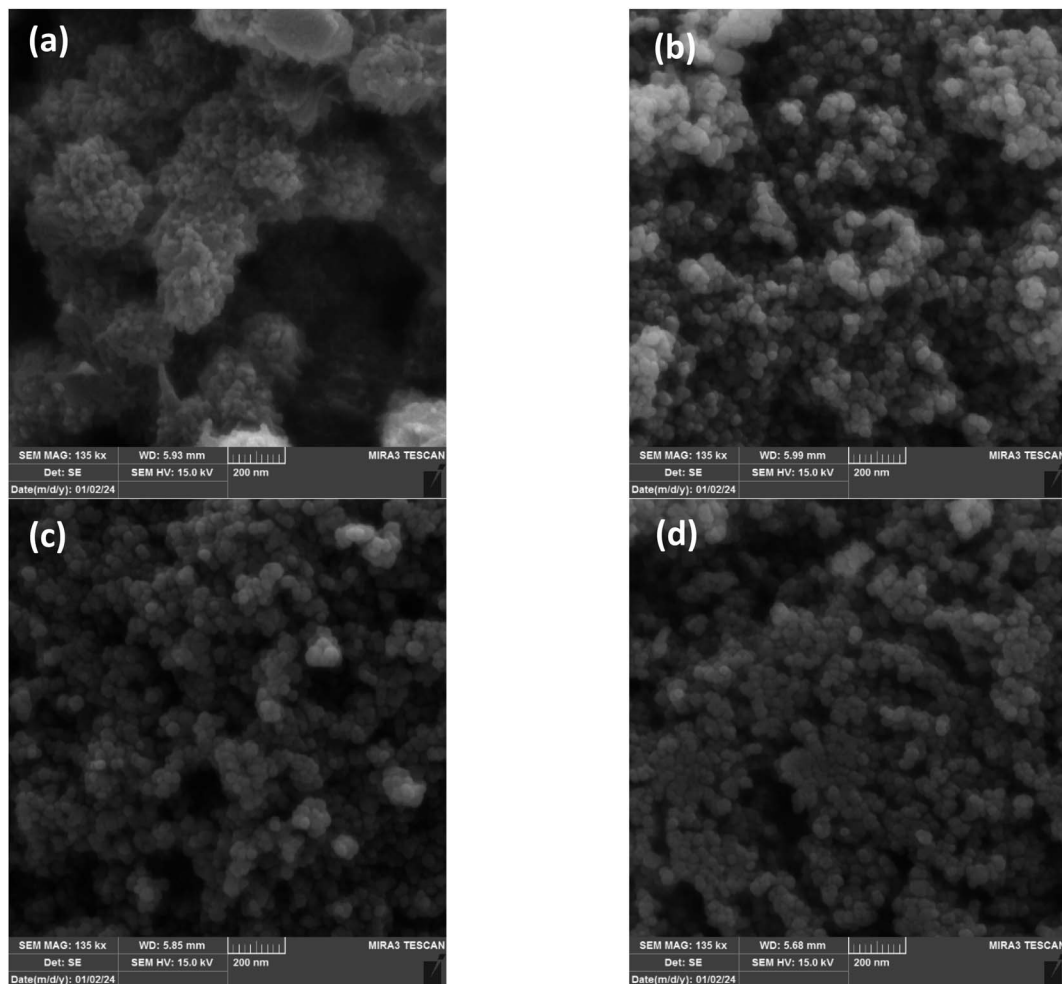


Fig. 2 FESEM image of (a) PANI, (b) 1 : 9, (c) 3 : 7, and (d) 4 : 6 NiO<sub>x</sub>/PANI.



332 nm with a slight blue shift is observable, and a shoulder peak around 377 nm, which affirms the successful formation of NiO<sub>x</sub>/PANI. The TEM images of NiO<sub>x</sub> nanoparticles are shown in Fig. 1d. The morphology of NiO<sub>x</sub> nanoparticles is spherical, with an approximate diameter of about 40 nm.

The morphology of PANI and NiO<sub>x</sub>/PANI (1 : 9, 3 : 7, and 4 : 6) was investigated using scanning electron microscopy (SEM) and shown in Fig. 2. Fig. 2a illustrates that the PANI nanofiber has a porous morphology, resulting in a larger surface area. The FESEM image of the NiO<sub>x</sub>/PANI indicates the incorporation of NiO<sub>x</sub> in the polymer matrix, with an average diameter in the nanometer range, likely due to the  $\pi$ - $\pi$  and electrovalent interactions between the PANI molecules and NiO<sub>x</sub>.<sup>39,40</sup> EDX-mapping analysis (Fig. 3) confirmed the successful formation of the nanocomposite, with well-distributed C, N, O, and Ni elements. In Fig. S3,† the elemental composition of the NiO<sub>x</sub>/PANI materials was also studied by EDX, with no trace of impurities observed. Fig. 4a-c show the synthesis of PANI NPs, NiO<sub>x</sub>-PANI, and NiO<sub>x</sub> NPs respectively. Complete details of device fabrication is given in ESI.†

The devices were also assembled with the following configuration: FTO/TiO<sub>2</sub>/FAPbI<sub>3</sub> perovskite/HTM doped carbon. *J*-*V*

measurements were used to evaluate the performance of PSCs under AM 1.5G irradiance (Fig. 5a). Based on the *J*-*V* curves, the photovoltaic parameters of the devices, including FF, open-circuit voltage (*V*<sub>OC</sub>), short circuit current density (*J*<sub>SC</sub>), and PCE, were calculated and are given in Table S1.† In general, the addition of NiO<sub>x</sub>/PANI as a HTM to carbon paste improves *J*<sub>SC</sub>, *V*<sub>OC</sub>, and FF parameters and raises device PCE. In which the device exhibited an improved PCE of 17.52%, along with a *V*<sub>OC</sub> of 1.114 V, a *J*<sub>SC</sub> of 23.00 mA cm<sup>-2</sup>, and an FF of 68.39. Notably, the PCE was further enhanced to 18.30% when NiO<sub>x</sub>/PANI was doped within the carbon electrode.

To determine the ideal weight ratio and study the effects of different NiO<sub>x</sub>/PANI weight ratios on photovoltaic performance, different perovskite solar cell amounts were made, and their *J*-*V* characteristics were assessed. The *J*-*V* curves of the best devices with various NiO<sub>x</sub>/PANI in carbon are displayed in Fig. 5a. Interestingly, mixing carbon paste with NiO<sub>x</sub>/PANI (3 : 7) yields the best results. Fig. 5b shows the *J*-*V* curve for the best-performing device under AM 1.5G illumination without HTM (control), NiO<sub>x</sub>, and PANI, as well as the device using NiO<sub>x</sub>/PANI (3 : 7) as HTM in carbon-based devices. Table S1† lists the

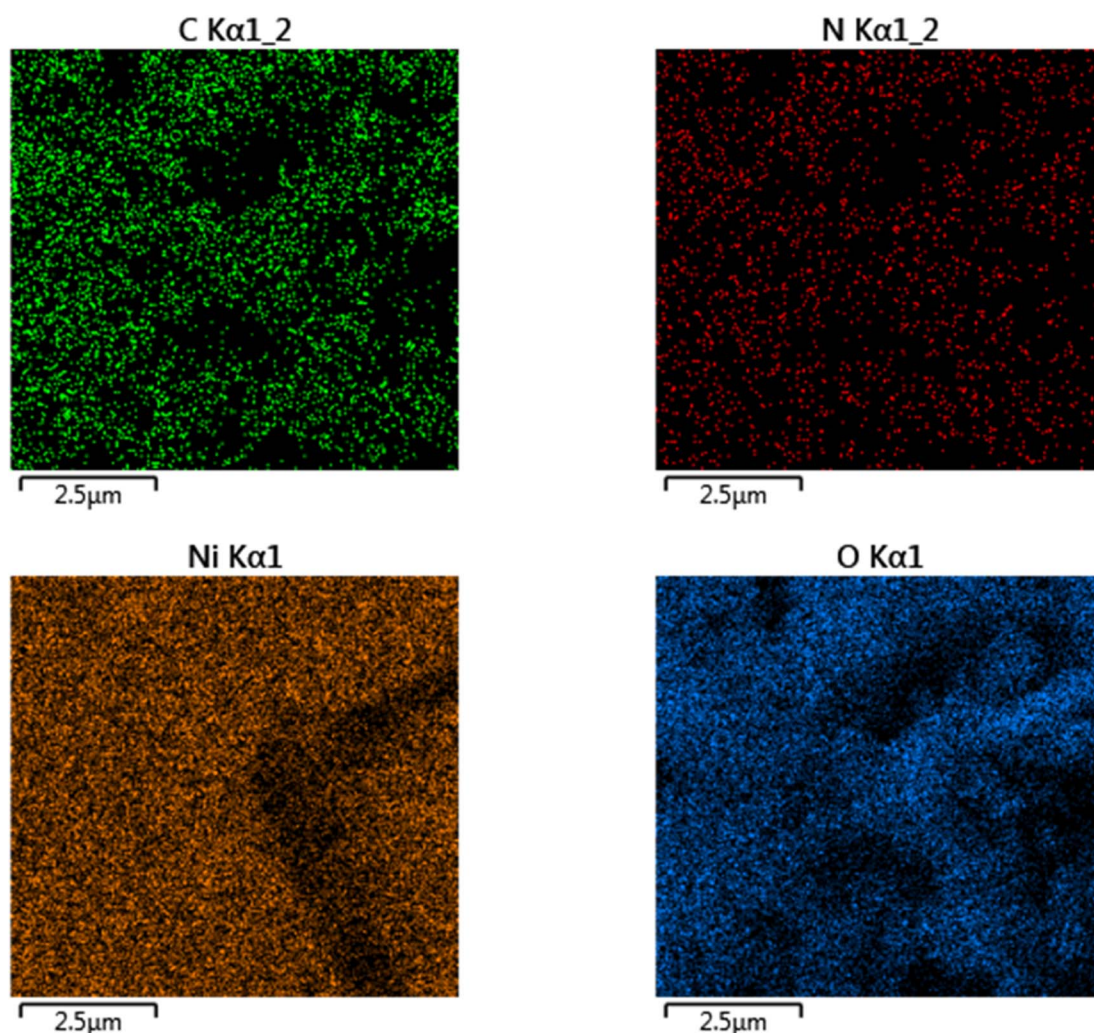


Fig. 3 Mapping FESEM to show elements distribution of C, N, Ni, and O on the NiO<sub>x</sub>/PANI nanocomposite.



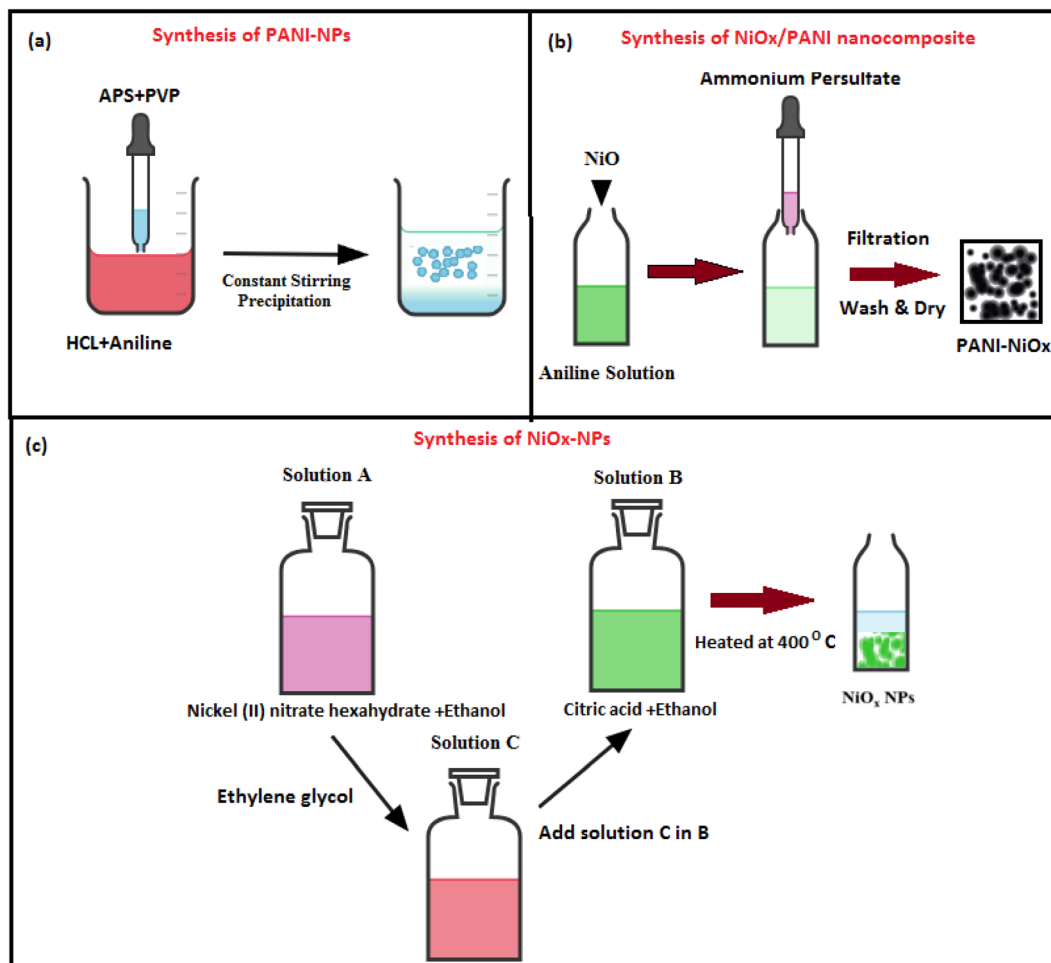


Fig. 4 Synthesis of (a) PANI-NPs (b) NiO<sub>x</sub>/PANI nanocomposite and (c) NiO<sub>x</sub> NPs.

associated photovoltaic parameters that were taken from the  $J$ - $V$  curve. The device incorporating NiO<sub>x</sub>/PANI (3 : 7)+CE HTM exhibits a notable enhancement in  $V_{OC}$  and  $J_{SC}$ . The integration of NiO<sub>x</sub>/PANI (3 : 7)+CE HTM is responsible for the increase in  $V_{OC}$  because it shifts the Fermi level of carbon downward.

It is worth mentioning that the inclusion of NiO<sub>x</sub>+carbon results in inferior device performance, despite having a slightly higher  $V_{OC}$  compared to a device without a hole transport material (HTM). The reduced efficiency can be attributed to the limitation in charge transport caused by the inadequate conductivity of NiO<sub>x</sub> HTM. Conversely, the introduction of NiO<sub>x</sub>/PANI greatly enhances the transfer of holes from the light-harvesting layer to the carbon electrode.

Fig. 5c displays the incident photon-to-electron conversion efficiency (IPCE) responses and integrated short circuit current density ( $J_{SC}$ ) for the control cell and perovskite solar cells (PSCs) utilizing NiO<sub>x</sub>/PANI(3 : 7)+CE as the hole transport material (HTM) and counter electrode. The results demonstrate that the PSC incorporating NiO<sub>x</sub>/PANI(3 : 7)+CE exhibits a distinct IPCE spectrum shape and higher IPCE value within the 350 to 900 nm range compared to devices utilizing pure carbon. Consequently, the integrated  $J_{SC}$  values of 22.74 and 22.80 mA cm<sup>-2</sup> align with

the  $J$ - $V$  characteristics of the devices. Fig. 5d illustrates the standard  $J$ - $V$  curves of PSCs fabricated with two types of cells: the control cell and PSCs based on NiO<sub>x</sub>/PANI(3 : 7)+CE. In both the forward and reverse scan directions, the  $J$ - $V$  curves were measured. The hysteresis problem of the current-voltage curve of the HTL-free carbon-based PSCs, in particular, can also be resolved by the NiO<sub>x</sub>/PANI. The electrical properties of the cells acquired from both the forward scan (FS) and reverse scan (RS) directions are shown in Table 1. Eqn (1) was utilized to evaluate the hysteresis changes and determine the hysteresis index (HI), which represents the degree of hysteresis.

$$HI = \frac{PCE_{rs} - PCE_{fs}}{PCE_{rs}} \quad (1)$$

In the equation,  $PCE_{rs}$  represents the calculated PCE from the reverse scan, while  $PCE_{fs}$  represents the calculated PCE from the forward scan. Larger HI values correspond to increased hysteresis. As depicted in Table 1, the control device exhibits the highest HI value at 10.28%. In contrast, the device incorporating NiO<sub>x</sub>/PANI showcases a lower HI value of 6.95%. The significant hysteresis observed in the control device is attributed to the poor interface between the light-harvester layer



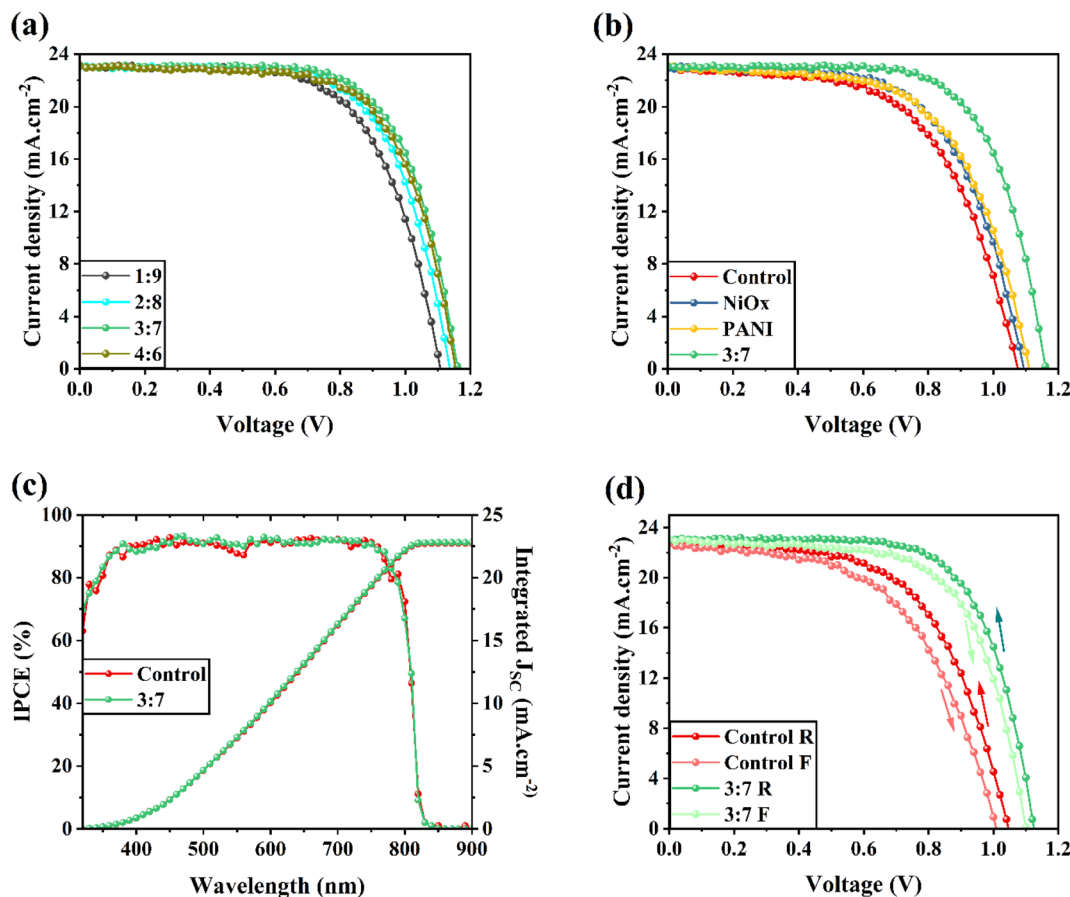


Fig. 5  $J$ - $V$  curves of different HTL-free CPSCs with carbon electrodes contained different  $\text{NiO}_x/\text{PANI}$  nanocomposites. (b)  $J$ - $V$  curves of devices with pure carbon electrode (control), carbon electrodes contained  $\text{NiO}_x$ , PANI, and  $\text{NiO}_x/\text{PANI}$  (3 : 7). (c) IPCE curves of control and 3 : 7 carbon electrode-based devices. (d)  $J$ - $V$  curves of control and 3 : 7 devices in forward and reverse directions.

Table 1 PCE of control and 3 : 7 based HTL-free CPSCs in reverse and forward sweeping directions to measure hysteresis index

Device name	PCE reverse (%)	PCE forward (%)	HI (%)
Control	14.02	12.58	10.28
3 : 7	17.75	16.52	6.95

and the carbon electrode, which negatively affects carrier transport through the interface. On the other hand, the  $\text{NiO}_x/\text{PANI}$ -based devices demonstrate improved interfacial contacts, resulting in reduced hysteresis and enhanced carrier transport.

In this study, we conducted a comparison of the stability between the device based on  $\text{NiO}_x/\text{PANI}$  and the control cell. The intrinsic affinity of  $\text{FAPbI}_3$  perovskites to reaction with moisture poses a challenge to the stability of PSCs, making it a primary concern. To evaluate the stability, the cells were exposed to ambient air with a relative humidity of 30–40% at room temperature in dark conditions, without encapsulation. Fig. 6a illustrates the stability of the cells over a period of 2400 hours. The PSC incorporating  $\text{NiO}_x/\text{PANI}+\text{CE}$  maintained 96% of its initial power PCE, while the control device retained 93% of its original performance. Furthermore, Fig. 6b displays the

stability of the PSCs under simulated sunlight irradiance at a humidity level of 25–30% and room temperature. The graph indicates that PSCs utilizing the  $\text{NiO}_x/\text{PANI}$  exhibit higher stability compared to the control PSCs. This enhanced stability can be attributed to the small porosity of  $\text{NiO}_x/\text{PANI}+\text{carbon}$ , resulting from the smaller size of the metal oxide on the carbon material and thick  $\text{NiO}_x/\text{PANI}+\text{CE}$  coating ( $\sim 15 \mu\text{m}$ ). The long-term stability performance of the  $\text{NiO}_x/\text{PANI}$ -based devices may be attributed to this decreased porosity, which prevents oxygen and moisture from penetrating through the counter electrode.

Next, we use the PSC's capacitance–voltage ( $C$ - $V$ ) curve to measure the carrier concentration. The Mott–Schottky equation ( $M$ - $S$ ) is used to calculate the carrier density based on the curve's slope. To determine the built-in potential ( $V_{\text{bi}}$ ), the linear portion of the Mott–Schottky plot in Fig. 7a was utilized. The  $V_{\text{bi}}$  of the  $\text{NiO}/\text{PANI}$  device was higher than that of the control device. This indicates that a high  $V_{\text{OC}}$  is produced by the  $\text{NiO}_x/\text{PANI}$ 's ability to increase charge separation and decrease carrier rearward movement at the carbon–perovskite interface.<sup>41</sup> Fig. 7b presents the impedance spectroscopy spectra obtained from various cells: the control cell, PSCs based on  $\text{NiO}+\text{CE}$ ,  $\text{PAN}+\text{CE}$ , and  $\text{NiO}_x/\text{PANI}(3:7)+\text{CE}$ . The measurements were



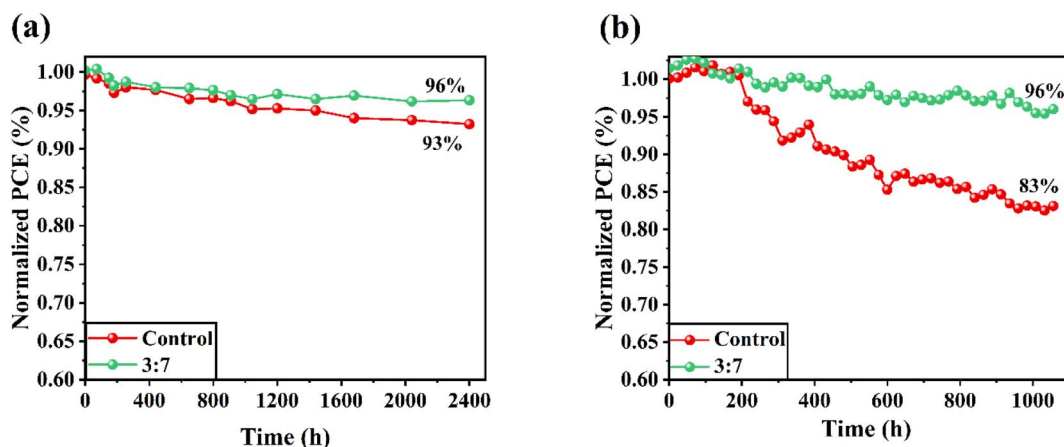


Fig. 6 Stability tests of control and 3:7-based HTL-free CPSCs against (a) ambient air with relative humidity of 30–40% in dark conditions at room temperature and (b) simulated sunlight irradiance at a humidity level of 25–30% at room temperature.

performed under dark conditions at open-circuit voltage. The inset of Fig. 5b showcases the equivalent circuit diagram that was utilized to fit the EIS curves. The charge recombination resistance ( $R_{\text{rec}}$ ) in the low-frequency region exhibited a negative correlation with the degree of charge recombination. Notably, the  $\text{NiO}_x/\text{PANI}+\text{CE}$  cell displayed a significantly higher  $R_{\text{rec}}$  value

(1687  $\Omega$ ) in comparison to the  $\text{NiO}+\text{CE}$  cell (897  $\Omega$ ), the  $\text{PANI}+\text{CE}$  cell (954  $\Omega$ ), and the control cell (784  $\Omega$ ). This indicates that the addition of  $\text{NiO}_x/\text{PANI}$  effectively suppressed charge carrier accumulation at the interfaces and reduced charge carrier recombination. These findings align with the results obtained from photoluminescence (PL) measurements. A device was

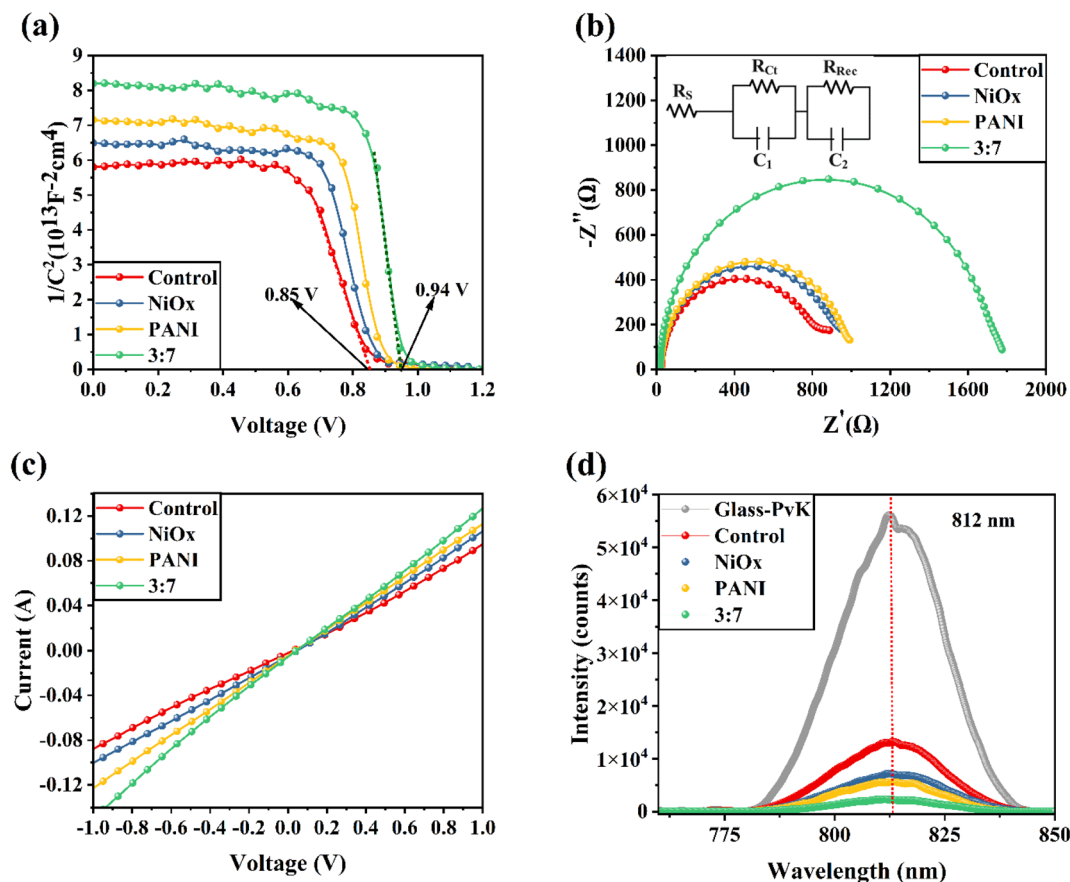


Fig. 7 (a) Mott–Schottky and (b) Nyquist plots of HTL-free CPSCs fabricated based on different carbon electrodes. Inset of Fig. 5b shows the equivalent circuit diagram used to fit EIS curves. (c) Dark  $I$ – $V$  of different carbon electrodes. (d) PL response of  $\text{FTO}/\text{ETL}/\text{FAPbI}_3/\text{different carbon electrodes}$ .



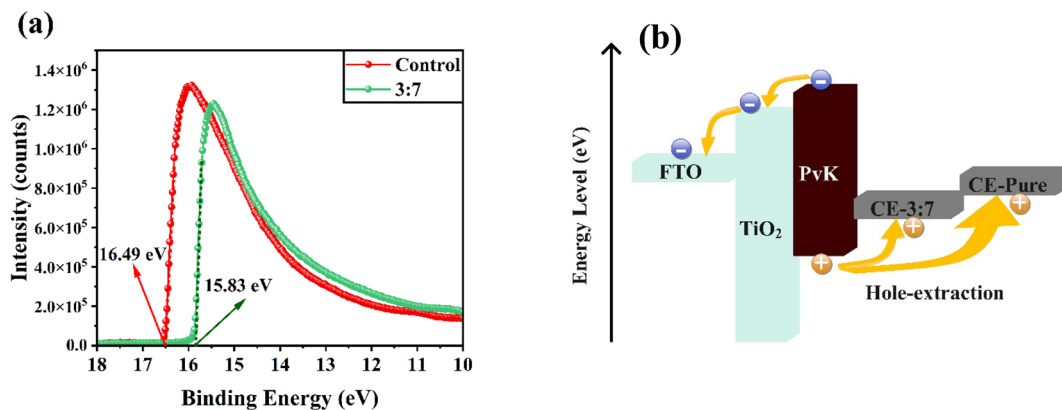


Fig. 8 (a) UPS spectra of control and 3:7 NiO<sub>x</sub>/PANI-contained carbon electrodes. (b) Energy band diagram of each component in the fabricated HTL-free CPSCs with control and 3:7 NiO<sub>x</sub>/PANI-contained carbon electrodes, showing the separation and transport of photo-generated carriers.

fabricated with a different carbon electrode architecture in order to study the vertical charge transport properties. As expected, the  $I$ - $V$  curve of the device containing a NiO<sub>x</sub>/PANI+CE showed an enhanced slope when compared to the control devices NiO<sub>x</sub>+CE and PANI+carbon electrode, as shown in Fig. 7c. The improved conductivity observed in the NiO/PANI+CE device can be attributed to its enhanced ability to extract holes from the perovskite film. This increase in conductivity aligns with the superior performance of the perovskite solar cells (PSCs), as demonstrated by the higher  $J_{SC}$  and FF.<sup>42</sup> Furthermore, the linear relationship observed in the  $J$ - $V$  curves indicates the presence of good ohmic contacts in all device configurations.<sup>43,44</sup>

To further validate the improved charge transport properties resulting from the integration of NiO/PANI into the carbon electrode (CE), Fig. 7d presents the steady-state photoluminescence (PL) spectra. The spectra compare a perovskite absorber film deposited on a glass/fluorine-doped tin oxide (FTO) substrate without any additional layers, with the structures FTO/ETL/FAPbI<sub>3</sub>/using different carbon electrodes. In the absence of additional layers, the bare perovskite film exhibited a distinct photoluminescence (PL) peak at 812 nm. The steady-state PL spectra provide clear evidence that the NiO<sub>x</sub>/PANI+CE configuration effectively suppresses the PL intensity, suggesting enhanced charge transfer from the light-harvesting layer.<sup>45,46</sup> This confirms that the addition of NiO<sub>x</sub>/PANI enhances the charge transport properties of the device.

To examine the band alignment between the NiO<sub>x</sub>/PANI+CE material and the perovskite absorber layer, ultraviolet photoelectron spectroscopy (UPS) test were measured. Fig. 8a displays the UPS spectra for pure carbon and NiO<sub>x</sub>/PANI+CE. The results indicate that the incorporation of NiO<sub>x</sub>/PANI has a minimal effect on the band positions. There is a shift of 0.66 eV in the work function (WF). The energy level at  $-5.39$  eV for NiO<sub>x</sub>/PANI+CE corresponds to an intermediate and favorable position, which facilitates efficient hole transfer from perovskite to the NiO<sub>x</sub>/PANI+CE (Fig. 8b).

### 3. Conclusion

In our research, we implemented a doping technique that involved incorporating a NiO<sub>x</sub>/PANI into the carbon electrode to increase the photovoltaic properties of PSCs. The introduction of this nanocomposite resulted in a beneficial energy band alignment with both the perovskite and carbon materials, leading to decreased resistance and improved conductivity. The unique structure of the nanocomposite provided a larger specific surface area compared to regular carbon, further facilitating the reduction of resistance and improvement of conductivity. Furthermore, the inclusion of the nanocomposite facilitated enhanced interface contact between the carbon electrode and the perovskite layer, resulting in reduced defect generation. The primary objective of incorporating the nanocomposite was to improve charge transfer, energy level alignment, and interface contact while also mitigating surface degradation of the perovskite. These combined effects effectively reduced charge recombination and minimized energy loss in the system. As a result, the overall efficiency increased to 18.30%, which is 26.5% higher than that of devices using a pure carbon electrode. Additionally, the NiO/PANI+CE-based PSCs exhibited an encouraging 96% of initial power conversion efficiency (PCE) after 2400 hours in an ambient atmosphere. The successful utilization of NiO/PANI+CE as the hole transport material and carbon electrode significantly contributed to the stability test's positive outcome.

### Conflicts of interest

The authors declare no conflict of interest.

### Acknowledgements

The authors present their appreciation to King Saud University for funding this research through Researchers Supporting Program number (RSPD2024R990), King Saud University, Riyadh, Saudi Arabia.



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