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# Fluorine-free nanoparticle coatings on cotton fabric: comparing the UV-protective and hydrophobic capabilities of silica vs. silica-ZnO nanostructures

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Robust, hydrophobic woven cotton fabrics were obtained through the sol-gel dip coating of two different nanoparticle (NP) architectures; silica and silica-ZnO. Water repellency values as high as 148° and relatively low tilt angles for fibrous fabrics (12°) were observed, without the need for fluorinated components. In all cases, this enhanced functionality was achieved with the broad retention of water vapor permeability characteristics, *i.e.*, less than 10% decrease. NP formation routes indicated direct bonding interactions in both the silica and silica-ZnO structures. The physico-chemical effects of NP-compatibilizer (*i.e.*, polydimethoxysilane (PDMS) and *n*-octyltriethoxysilane (OTES) at different ratios) coatings on cotton fibres indicate that compatibilizer-NP interactions are predominantly physical. Whenever photoactive ZnO-containing additives were used, there was a minor decrease in hydrophobic character, but order of magnitude increases in UV-protective capability (*i.e.*, UPF > 384); properties which were absent in non-ZnO-containing samples. Such water repellency and UPF capabilities were stable to both laundering and UV-exposure, resisting the commonly encountered UV-induced wettability transitions associated with photoactive ZnO. These results suggest that ZnO-containing silica NP coatings on cotton can confer both excellent and persistent surface hydrophobicity as well as UV-protective capability, with potential uses in wearables and functional textiles applications.

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## 1. Introduction

Surface functionalization of textiles for hydrophobic (*i.e.*, water-repellent) and ultraviolet (UV)-protective capability is increasingly important for high-performance technical textiles, outdoor wear and protective fabrics applications.<sup>1,2</sup> Cellulosic cotton is the most widely used natural fibre in the textile industry and comprises numerous surface hydroxyl groups, which renders such substrates intrinsically hydrophilic and hygroscopic.<sup>3-5</sup> Thus, effective, durable and stable functionalization routes for controlling cotton wettability are of great interest.

Organosilicon and fluoropolymer-based compounds are reliably used to fabricate superhydrophobic textile surfaces while also keeping fabrics relatively soft and breathable.<sup>6,7</sup> Perfluorinated compounds (PFC), depending on their structure, can yield highly water-repellent surfaces due to their ability to lower surface energy below that of any other common chemical treatment.<sup>8</sup> However, PFC are often expensive, environmentally persistent – over a decade in some cases – and pose potential

human-health issues, *e.g.*, can interfere with reproductive and endocrine systems as well as stimulate tumor growth.<sup>9,10</sup> Therefore, fluorine-free, low cost and environment-benign hydrophobic technologies are in favor.<sup>11,12</sup> Silica-based, and zinc oxide (ZnO)-based coatings are popular alternatives.

Surface wetting is determined by static contact angle measurements; hydrophobic surfaces exhibit a contact angle between a droplet and surface exceeding 90° (superhydrophobic when above 150°), which often also results in a low sliding angle, below 10°, although this is often higher for fibrous substrates with complex, irregular microstructures.<sup>13,14</sup> Conferrence of surface (super-)hydrophobicity usually requires: (i) modification for a low surface energy and; (ii) a hierarchical nano/micron-scale topographical roughness (to minimize contact areas between a liquid droplet and the solid surface).<sup>15,16</sup> Both conditions, as well as any electrostatic interactions, decrease the sliding resistance and so, tilt angle of an impingent water droplet, based on differing liquid–solid lateral adhesion forces in both the static and kinetic regimes, reflecting improved water repellency.<sup>17,18</sup>

Silica-based coatings fulfil the first requirement of surface hydrophobicity, while offering low environmental and bio-accumulation risks, as well as being durable; resistant to

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mechanical abrasion, and stable to liquids exposure.<sup>19,20</sup> However, inorganic PFC-alternatives best satisfy the second hydrophobic condition, and also possess excellent robustness, *e.g.*, photoactive ZnO nanoparticles (NP) present surface-roughened hydrophobicity and can also absorb UV-radiation – beneficial properties that silica alone cannot provide.<sup>21–23</sup> Such ZnO UV-shielding capabilities have been used both in cosmetics and wearables.<sup>24,25</sup> Fabric UV-blocking is characterized by the UV protective factor (UPF), which protects against both UV-A (320–400 nm) and UV-B (280–320 nm) radiation sources to *e.g.*, erythema, among other issues.<sup>26,27</sup> Such optically-protective coatings help surfaces resist UV-photodegradation, *e.g.*, reduced photo-yellowing of wool.<sup>28,29</sup> However, an issue with ZnO (and other photoactive metal oxides) is that UV-exposure, even briefly, often causes a light-induced transition, *e.g.*, UV-induced superhydrophilicity; transforming from superhydrophobic to superhydrophilic, or *vice versa*.<sup>30,31</sup> Nano-composite structures that resist such transitions would allow for reliably consistent functional properties over time – a critical requirement for many commercial applications.

The objective of this work was to compare silica NP and silica-ZnO NP coated substrate effects, and show that ZnO presence affords dual UV-protective and water-repellency properties, while also being resistant to UV-induced changes in surface wettability. Although similar architectures have been reported for drug delivery, bio-imaging, photonics and other applications, such combined silica-ZnO structures are less well-explored for textile functionalization.<sup>32–34</sup> Both structures are also fluorine-free, relatively benign in nature, have proven capability, are cost-efficient, and separately, are already ubiquitous in wearables applications.<sup>35</sup>

Silane-type coupling agents act as interfacial adhesives between an inorganic material and organic substrates, *e.g.*, in textiles, enhancing loading, dispersion and binding.<sup>36,37</sup> Poly(dimethyl-siloxane) (PDMS) and *n*-octyltriethoxysilane (OTES) – two commonly used long-chain alkoxide hydrophobizing agents – were used at different loadings in this study to help graft the NP to substrates and so, enhance the overall hydrophobic surface properties.<sup>38–40</sup> Absent the use of such agents, the durability of NP coatings on textiles is greatly diminished. We identified which compatibilizer was most effective for hydrophobic and UPF applications respectively.

A directly-controllable sol–gel dip coating approach was used on fabrics to impart roughness and increase hydrophobicity, *via* loading of pre-formed NP. This deposition method does not need salt, electrolytes, or extended high-temperature processing, unlike other common coating techniques.<sup>41–43</sup> The resultant composites were observed for differences in hydrophobicity and UPF functionality, as well as the relative stability of such fabric coatings to UV-light exposure and extended laundering treatments.

## 2. Methodology

### 2.1 Chemicals

Chemicals used as-received were; acetone (VWR); ammonia, (VWR, 28–32%); ethanol absolute (EtOH, Unichem, 99.5%); *n*-

octyltriethoxysilane (OTES, Alfa Aesar, 95%); poly(dimethyl-siloxane) (PDMS, Sigma Aldrich,  $M_w = 95\,000$ ); sodium hydroxide (NaOH, International Laboratory, USA); tetraethylorthosilicate (TEOS,  $(\text{Si}(\text{OC}_2\text{H}_5)_4)$ , Sigma-Aldrich, 98%); toluene (VWR-Chem AR); triethanolamine (TEA, Xilong Chemical, 98%); zinc acetate dihydrate ( $\text{Zn}(\text{OAc})_2$ , Acros Organic) and deionized (DI) water (18 M $\Omega$ ).

### 2.2 Substrate

Woven, bleached, pre-washed cotton fabric substrates of  $1 \times 1$  inch ( $2.54 \times 2.54$  cm  $\times$   $\sim 0.2$  mm;  $l \times w \times t$ ) were used for deposition. To remove any impurities, substrates were cleaned with a non-ionic aqueous detergent mixture in a washing machine at 85 °C for 15 min, then rinsed with DI water.

### 2.3 Two-step synthesis

(1) Preparation of silica and silica-ZnO NP: silica NP were prepared *via* the Stöber method comprising hydrolysis then subsequent, or even concurrent, condensation.<sup>44,45</sup> TEOS (25 ml; silica precursor), water (75 ml) and  $\text{NH}_3 \cdot \text{H}_2\text{O}$  (25 ml; catalyst) were mixed under stirring, then aged for 4 h.<sup>46</sup> Products were collected by centrifugation and washed three times with ethanol, then oven-dried. To prepare silica-ZnO composites, a simplified controlled chemical double jet precipitation technique was used by adding TEA precipitating agent and  $\text{Zn}(\text{OAc})_2$  precursor into a silica-ethanol aqueous solution simultaneously.<sup>47,48</sup> Briefly, silica (0.2 g) was dispersed in a 30 mL EtOH–water solution and the mixture heated to 90 °C. After 10 min, 10 ml of  $1.6 \text{ mol L}^{-1}$  TEA and 20 ml of  $0.02 \text{ mol L}^{-1}$   $\text{Zn}(\text{OAc})_2$  solution were dropped simultaneously into the silica-EtOH aqueous solution at a constant flow rate. The system was then stirred continuously at 90 °C for 1 h. The resulting white powders were centrifuged, washed repeatedly with distilled water, and oven-dried. Finally, the powders were sintered for 3 h at 700 °C to remove the TEA, resulting in the silica-ZnO NP material. Such formation routes yield Si–O–Si bonding structures, as well as points of attachment that survive hydrolysis and condensation. (2) Coating NP onto cotton fabric substrates. Substrates were treated with the solution containing PDMS, OTES and the NP solution. First, PDMS (1.9 g) was dissolved in toluene (48 mL) and stirred for 1 h at room temperature. Then, the weight percentage concentration of 2% or 86% of OTES (relative to PDMS) was added with a further 3 h stirring at room temperature. After this, the pre-prepared NP powder (0.1 g) was ultrasonically dispersed in solution for 30 min. Subsequently, the cotton fabric was dip-coated in this mixture; a 20 min immersion period followed by withdrawal ( $1 \text{ cm s}^{-1}$ ), followed by a short evaporation, and then curing at 140 °C for 1 h to aid solvent removal, and allow for full densification of the oxide matrix.

### 2.4 UV-exposure testing

An artificial light source, UV lamp (Philips, 18 W) emitting a gauss shaped spectrum that peaked at 370 nm and with cut off at 340 nm and 400 nm respectively was used. Samples were placed under the UV lamp at a substrate-lamp distance of 16 cm, for up to 40 h.



## 2.5 Characterization

Nanoparticle formation was probed *via* time-resolved *in situ* analysis of reaction progress. This allowed for empirical determination of reaction mechanisms. In such cases, the desired chemicals were mixed at known ratios immediately prior to loading onto sample stages for analysis, and data acquisition done while time elapsed was recorded.<sup>49</sup> The morphologies and microstructures of pre-formed NP and fabric-coated samples were examined using a TESCAN VEGA3 tungsten thermionic emission scanning electron microscope (SEM) on gold sputter coated samples. Energy dispersive X-ray analysis (EDS) was also carried out along with SEM. Transmission electron microscopy (TEM) images were acquired on a JEOL JEM-2100F, inclusive of EDS elemental mapping, on gold-sputtered samples. X-ray diffraction (XRD) was used for determining crystallinity and phase purity on a Rigaki SmartLab, operated at 40 kV and 100 mA with Cu-K $\alpha$  radiation in a Bragg–Brentano configuration and 5° divergence, acquired over the 15–90°  $2\theta$  range, at a 0.02° step size. UV-visible spectra of pre-formed NP mixtures were acquired on a Hitachi Double Beam Spectrophotometer UH5300, against a DI H<sub>2</sub>O standard over a 250–750 nm range. Analysis of pre-formed NP mixtures using dynamic light scattering (DLS) technique was conducted on a Zetasizer nano-ZS (Malvern Panalytical, Malvern, UK) with a backscattering detection angle of 173° and a thermostat with 60 s thermal equilibration time.<sup>50</sup> The NP were randomly dispersed in DI-H<sub>2</sub>O to determine the hydrodynamic diameters; solutions were injected into the sample cuvette that was rinsed with Milli-Q water twice and dried before use. Temperatures were maintained at room temperature during the analysis. The hydrodynamic diameter (z-average in nm) was determined from the translational diffusion coefficient, using the Stokes–Einstein equation. All measurements were taken three times and averaged.

Coated fabrics were analyzed using transmission ATR-FTIR on a PerkinElmer Spectrum 100 over 4000–650 cm<sup>-1</sup> at 4 cm<sup>-1</sup> resolution and 8 scans. Color strength, K/S data, was acquired on a Macbeth ColorEye (CE-7000A) colorimeter, based on D65 illuminant and 10° observer values, with a small aperture view, across 400–700 nm, with the UV filtered and the specular component excluded, across four measurements for each sample to help minimize fibre orientation effects. A SCA 20, Data-Physics Instrument video optical contact angle goniometer was used to measure surface wettability of cotton substrates, with a 15  $\mu$ L DI water droplet size for all samples delivered from a needle connected to a syringe pump, at 24  $\pm$  0.5 °C and relative humidity of 40  $\pm$  3%.<sup>51,52</sup> Liquid droplets were placed onto the sample surface and the lateral, static water contact angles (WCA) measured; data continuously acquired for a 20 s period after initial impingement. All WCA were determined by averaging values measured at multiple different points on repeat sample surfaces with a water droplet of 15  $\mu$ L. Because the protruding fibers have some elasticity and can thus exhibit forces on the water droplet, accurate values for advancing and receding WCA are hard to acquire, so only static WCA are reported. For sliding angle investigation *via* a tilted

drop method, a drop of water was placed on the sample surface then tilted at a known angle (at step sizes of 0.5°) and the point at which the liquid droplet begins to “roll” was recorded as the sliding angle. A lower sliding angle indicates greater ease of droplet movement across and down the fabric surface.

Water vapor permeability testing was done using a modified BS7209 method. After fabrics were placed in a conditioning room overnight (at 40  $\pm$  2 °C and 90%  $\pm$  5% RH), test samples were sealed over an open receptacle containing a known volume of water, and the assembly was placed in a controlled atmosphere at room temperature. After a 6 h pause to establish an equilibrium of water vapor pressure gradient across the sample, successive weightings of the fabric-covered assembly were made at set intervals, and the water vapor transmission rate (WVTR) was calculated to investigate the water vapor permeability using the following equation:

$$\text{WVTR} = 24M/At \quad (1)$$

where  $M$  is the mass loss of the assembly over the period of time,  $t$  (in g) is the time between successive weightings of the assembly (in h) and  $A$  is the test area of the exposed test fabric, equal to the internal area of the receptacle cups (m<sup>2</sup>). The units are gm<sup>-2</sup> 24 h.<sup>53</sup> WVTR is the industry standard for evaluation of water vapor permeability (*i.e.*, the capacity of substrates to resist moisture transmission through the fabric and out into the atmosphere) and is the focus of most investigations into waterproof breathable textiles. Lower values indicate higher moisture resistance, and trapped moisture which in turn causes wearer discomfort. Perspiration and evaporation allow the human body to cool down with a high WVTR and so, even waterproof fabrics must maintain a high level of breathability to ensure comfort.<sup>53,54</sup>

UV-visible absorbance data was acquired on a Varian Cary 300 Conc UV-vis spectrophotometer, over a 200–800 nm range at a step size of 0.5 nm. The UPF values were calculated from:

$$\text{UPF} = \frac{\sum_{290}^{400} E(\lambda) \times S(\lambda)}{\sum_{290}^{400} E(\lambda) \times S(\lambda) \times T(\lambda)} \times 100 \quad (2)$$

where  $E(\lambda)$  is the CIE reference erythema dose spectrum,  $S(\lambda)$  is the radiation intensity distribution of sunlight, and  $T(\lambda)$  is the diffuse transmittance spectrum (%).<sup>55,56</sup>

## 2.6 Fastness to domestic and commercial laundering test

Laundering (including washing, rinsing and drying of fabrics) commonly causes coated material loss. Thus, laundering durability testing replicates the various temperature, detergent alkalinity, bleaching and friction/grinding forces encountered during a standard wash cycle. Each accelerated wash cycle is equivalent to three standard real-world, domestic wash cycles. Wash durability was evaluated at 40 °C with the equivalent of 4 g/500 ml ECE Phosphate Reference Detergent (SDC Type 3) in water, for 30 min, with ten steel balls (each weighing  $\sim$ 0.9 g, with a diameter of  $\sim$ 0.7 mm), in a Launder-O-meter (Atlas, USA).



Subsequently, all fabrics were cleaned at 40 °C with 100 ml water for each sample, for 1 min. Three accelerated wash cycles were carried out on samples, (*i.e.*, equivalent to nine standard wash cycles), after which UPF and water repellency properties were retested.

## 3. Results

### 3.1 Reaction schemes and nanoparticle analysis

Sol-gel derived silica NP and silica-ZnO NP were used to fabricate water-repellent surfaces. The formation processes of NP were analyzed (prior to their coating onto substrates), in order to understand the reaction routes (Fig. 1a-d).<sup>49</sup> Stöber-based silica NP were made *via* an ammonia-catalyzed, hydrolysis-condensation polymerization.<sup>57</sup> Simply, TEOS reacts with hydroxide to form a silanol, which condenses to the formation of siloxide ions and water. A siloxane linkage subsequently forms through attack on the silanol silicon centre (the rate-

determining step), rendering a silane network on the NP surface.<sup>58,59</sup> The final FTIR spectra also indicates peaks spanning 1400–1200  $\text{cm}^{-1}$  which signal Si–O–Si bridging bonds *via* condensation; specifically, the Si–O–Si asymmetric stretch and the strained siloxane networks in amorphous silica.<sup>60</sup> The sharp peak around 1066  $\text{cm}^{-1}$  represent the oxygen asymmetric stretch which is obvious in both NPs FTIR spectra.<sup>61</sup> This is further confirmed by the appearance of a shoulder at 946  $\text{cm}^{-1}$  (and a weak signal at  $\sim 890 \text{ cm}^{-1}$ ) upon heat treatment.

ZnO addition was done *via* a double jet precipitation method under aqueous, basic conditions, with initial formation of  $\beta\text{-Zn(OH)}_2$ , followed by rapid transformation to ZnO – the ethanolic conditions allowing for smaller, well-dispersed NP sizes.<sup>62</sup>  $\text{Zn(OAc)}_2$  is decomposed into interstices of the inorganic silica network, resulting in Si–O–Zn and Si–O–Si bands, corresponding to a strong bonding interaction between ZnO and the silylating agent.<sup>63</sup> The FTIR peak near 1390  $\text{cm}^{-1}$  may represent a bond between the –OH groups of the ZnO surface and TEOS

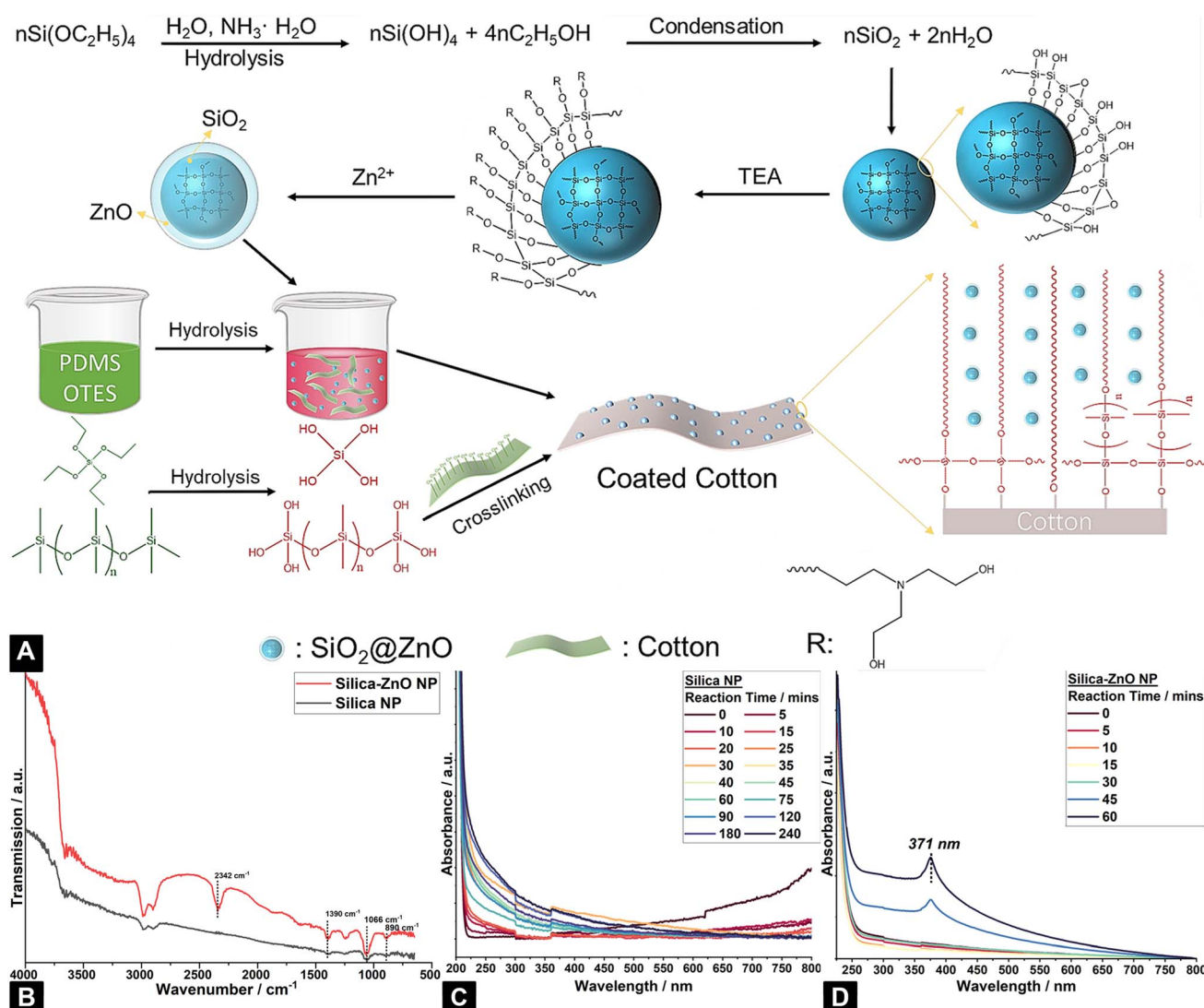


Fig. 1 (A) Reaction scheme for the nanoparticle formation and cotton fabric coating; (B) ATR-FTIR spectra of dried silica NP and silica-ZnO NP powder and, the time-resolved UV-vis analyses of silica NP (C) and silica-ZnO NP (D) formation.



carbonyl groups.<sup>64</sup> During this process, the silanol groups in siloxane oligomers may also react with TEA to subsequently aid further reactivity with  $\text{Zn}(\text{OAc})_2$  to form  $\text{ZnO}$  molecular clusters that can readily attach to the silica surface.<sup>65</sup> An obvious peak at  $\sim 2342\text{ cm}^{-1}$  in the silica-ZnO spectrum may correlate with trapped  $\text{CO}_2$  in the air ambient.<sup>66</sup>

Both NP size and distribution impact the agglomeration behavior of precursor mixtures and pre-formed materials, which in turn affects hydrophobic character. UV-vis analyses of the silica-ZnO NP mixtures yield a very weak SPR-based signal with a maximum at  $\sim 371\text{ nm}$ .<sup>67,68</sup> This is due to the ZnO band edge absorption peak.<sup>69,70</sup> No detectable signals were observed for the silica NP mixtures.

Preformed NP powders were also analyzed *via* XRD (Fig. 2a). For silica NP, the peak centred at  $2\theta \sim 24^\circ$  indicates a broadly amorphous structure, although calcination seems to have resulted in partial densification, agglomeration and a tendency towards crystallinity.<sup>71</sup> Conversely, silica-ZnO samples show peaks characteristic of hexagonal wurtzite ZnO, indicating a *c*-axis preferential orientation along the (002) direction, which corresponds well with the TEM data, and is often reported with low-temperature ZnO synthesis routes.<sup>72</sup> There is also a slight amorphous background signal between  $21$  and  $24^\circ$  that can be

attributed to silica presence. No other phases or impurities were detected. From DLS (Fig. 2b), the silica-ZnO and silica NP mean hydrodynamic radii were  $\sim 145\text{ nm}$  and  $\sim 225\text{ nm}$  (with particle count frequency ranges of  $130\text{--}150\text{ nm}$  and  $160\text{--}240\text{ nm}$ ), respectively. Also, single, narrow distribution curves were observed for both sample types, which indicates well-dispersed, uniform particle sizes without various aggregates. Both types of NP seemingly have relatively similar size distributions, with the silica-ZnO samples marginally smaller – the ZnO potentially inhibited agglomeration and Ostwald ripening of silica, while the effects of variant synthesis conditions also undoubtedly affect the reaction process, to impact particle size and distribution.<sup>73–75</sup>

TEM images of the pre-formed silica NP and silica-ZnO NP, along with their elemental analyses – measured across the respective particle clusters diameters *via* an EDS line scan function<sup>76</sup> – are shown in Fig. 2c and d. All samples display well-defined particles and related aggregates, with an acceptable dispersion for TEM observation. The mean silica particle size was  $\sim 420\text{ nm}$ ; the larger sizes as compared to DLS measurement perhaps a result of small particle agglomerates. The surface morphology of the silica matrix was atomically rough implying uneven thickness and no discernible lattice fringes

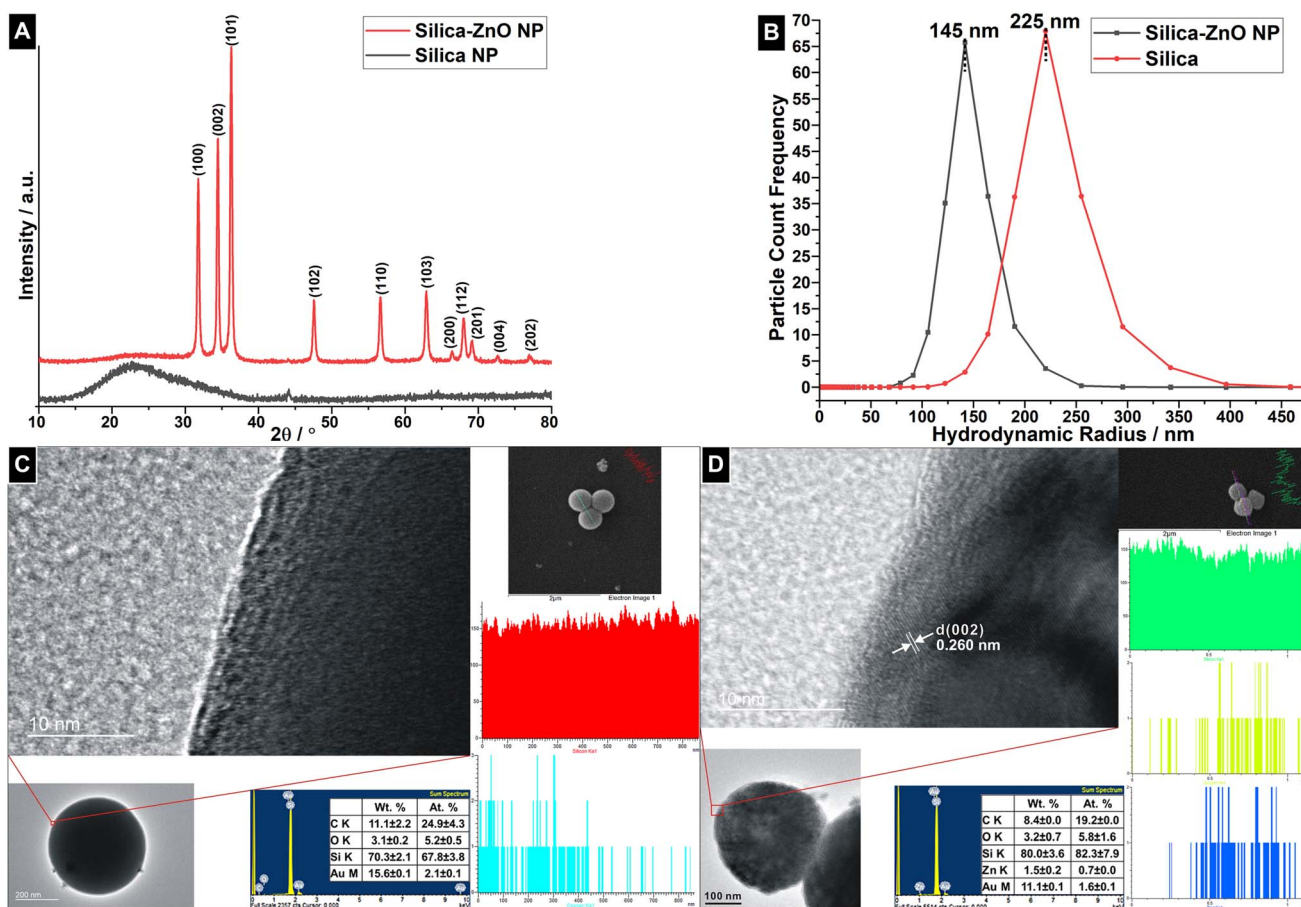


Fig. 2 (A) XRD analyzes of silica NP and silica-ZnO NP powders; (B) DLS comparison of silica NP and silica-ZnO NP mixtures, and; TEM and elemental line scan analyses (with averaged area scan elemental composition tables inset) of; (C) pre-formed silica NP powder and; (D) pre-formed silica-ZnO NP powder.



were observed, indicating that no inclusions were embedded (e.g., no ZnO lattice structures). Conversely, dark lattice spacing stripes were observed in silica-ZnO NP at 10 nm scale, with  $d$ -spacings of 2.60 Å solely discernible, consistent with the standard ZnO (002) crystalline plane of a hexagonal structure.<sup>77,78</sup> Elemental compositions indicate the sole presence of silica in the silica NP, with broad spectrum presence throughout the entire NP diameter.<sup>79</sup> Conversely, the silica-ZnO NP clearly exhibit Zn-M $\alpha$  element presence (usually concurrent with the O-K $\alpha$  signal), albeit at low concentrations and without any clear maxima or minima in the distribution observable.<sup>80</sup>

### 3.2 Coated fabric characterization and testing

Sol-gel dip coatings of the NP formulations were made using two hydrophobizing agents, *i.e.*, PDMS and OTES. These intrinsically water-repellent silicone thermoset polymers undergo covalent cross-linking to the fabric surface during curing, forming a rigid network of polymer chains that potentially stabilizes NP additives onto the woven cotton substrates; the microporous matrix seemingly stabilizing high particle dispersions.<sup>81,82</sup> Two different ratios of PDMS : OTES were used (*i.e.*, OTES at 2% and 86% loading by weight respectively), in order to observe the different effects of such binder/hydrophobizing agent towards hydrophobic properties as well as NP loading. This difference arises because PDMS terminates in trimethylsilyl groups whereas OTES contains non-polar C–C(H) groups; the latter should allow for higher NP-additive loading due to enhanced covalent cross-linking, potentially leading to additive-derived effects.<sup>83,84</sup> For the NP additives, the solvated zinc acetate precursor may initially bridge between the silica and PDMS, which, after heat treatment and/or extended UV exposure, the –O group from resultant tetrahedral ZnO forms hydrogen bonds with the –H group of PDMS (*i.e.*, Si–O–Zn linkages), enhancing the overall adhesion properties.<sup>85,86</sup> Alternatively, in the absence of such binding effects, the ZnO NP may form embedded nanostructures.<sup>87</sup>

Coated fabrics were used without further modification and exhibited non-wetting behavior for water, to different degrees, absent the need for fluorine-containing agents.<sup>88</sup> This is in stark contrast to the cotton reference which is intrinsically absorptive, and offers no discernible wettability data. The generally colorless NP afforded little color change to the substrates, aided by the sparseness of coating and low loadings used.<sup>89</sup> Coatings showed good adhesion to substrates, the resulting surfaces were stable to repeated wetting, rubbing and bending.

Fabric coated samples had no clearly discernible FTIR peaks, only a muting of the cellulose signals which increases with higher silica loading (Fig. 3a).<sup>90,91</sup> This indicates that incorporation and interaction between the NP and compatibilizers is predominantly a physical (not chemical) one. The CIE  $L^*a^*b^*$  color scale allows for tonal variations between samples to be identified (Fig. 3b and c). When compared to the reference cotton standard,  $L^*$ , (a more positive value indicating a lighter sample), shows a minor drop for all samples. The  $a^*$  values (more positive for a greater red contribution; a more negative for increasing green), show a positive increment with increased

material loading, with the highest value for NP inclusions. The  $b^*$  values (more positive for a greater yellow contribution; a more negative for increasing blue) are generally positive and similar for all except silica NP. Any addition to the substrate increases the color strength, K/S; the NP structures causing the greatest increase, although seemingly to similar extents, which is reasonable since both are composed of absorbing materials – the silica NP being marginally higher than the silica-ZnO. This is also expected due to the milky instead of pure white color shade of silica NP powders.<sup>92</sup>

Since hydrophobicity and its variation depends at least partly, on surface roughness, imaging was done on coated substrates to determine morphology. SEM images of the silica-based materials indicate clear presence of well-distributed, discontinuous coatings as surface irregularities due to particle clusters on cotton fibres (Fig. 3d). In all cases, a classical spherical-type structure is observed, of relatively uniform size (~0.5–2  $\mu\text{m}$ ), with little variation in morphology of granular protrusions between the different coating arrangements. Moreover, measurements seemingly correspond well to both particle size and TEM analyses. Further, fabric elemental analysis *via* EDS seems to corroborate the hypothesis that a higher relative amount of OTES results in a greater loading of material; there is significantly more zinc detected when more OTES was used.

From the static WCA as a function of time (Fig. 4a and b), all coated samples displayed hydrophobicity, as compared to an uncoated cotton fabric reference. On fibrous hydrophobic surfaces, water forms bead-like drops in order to reduce surface free energy, and this was especially the case for the NP-containing samples; WCA values reliably far in excess of the solely PDMS-OTES coating indicate that repellency enhancements are mainly due to NP addition, and are not predominantly silicone binder material based.<sup>93,94</sup> However, both sample conditions containing higher OTES (86%) presented an overall better hydrophobic character due to the hydrophobic nature of the OTES 8-carbon hydrocarbon chain.<sup>95</sup> With only the siliceous hydrophobizing agents, the lowest WCA values and the shortest latency periods before droplet diffusion through the substrate, was discovered for OTES(2%)-PDMS samples, in association with the highest obtained sliding angle of 19°.

Although certain individual samples showed superhydrophobic character (*i.e.*, water contact angle as high as 169° for OTES(2%)-PDMS silica-ZnO, albeit with the largest performance range), sample repeats averaged out to a reliably high hydrophobic. The most significant average WCA values were acquired from OTES(86%)-PDMS silica NP. Thus, cotton fabric was hydrophilic, PDMS-OTES coupling agents decreased the surface energy of the cotton fabric seemingly to improve hydrophobic characteristics, although this is not a long-lasting effect; impinging droplets readily pass through the porous substrate upon extended contact. Both NP structure additives created a rough surface that further increased the water repellency – slightly more so in the case of silica – to enhance the, likely Cassie-Baxter-type behavior (*i.e.*, the liquid makes partial contact balancing on the elevated regions of the solid, while also resting partly upon air pockets).<sup>96</sup> Moreover, the NP additives seem to stabilize fabrics to water penetration over time



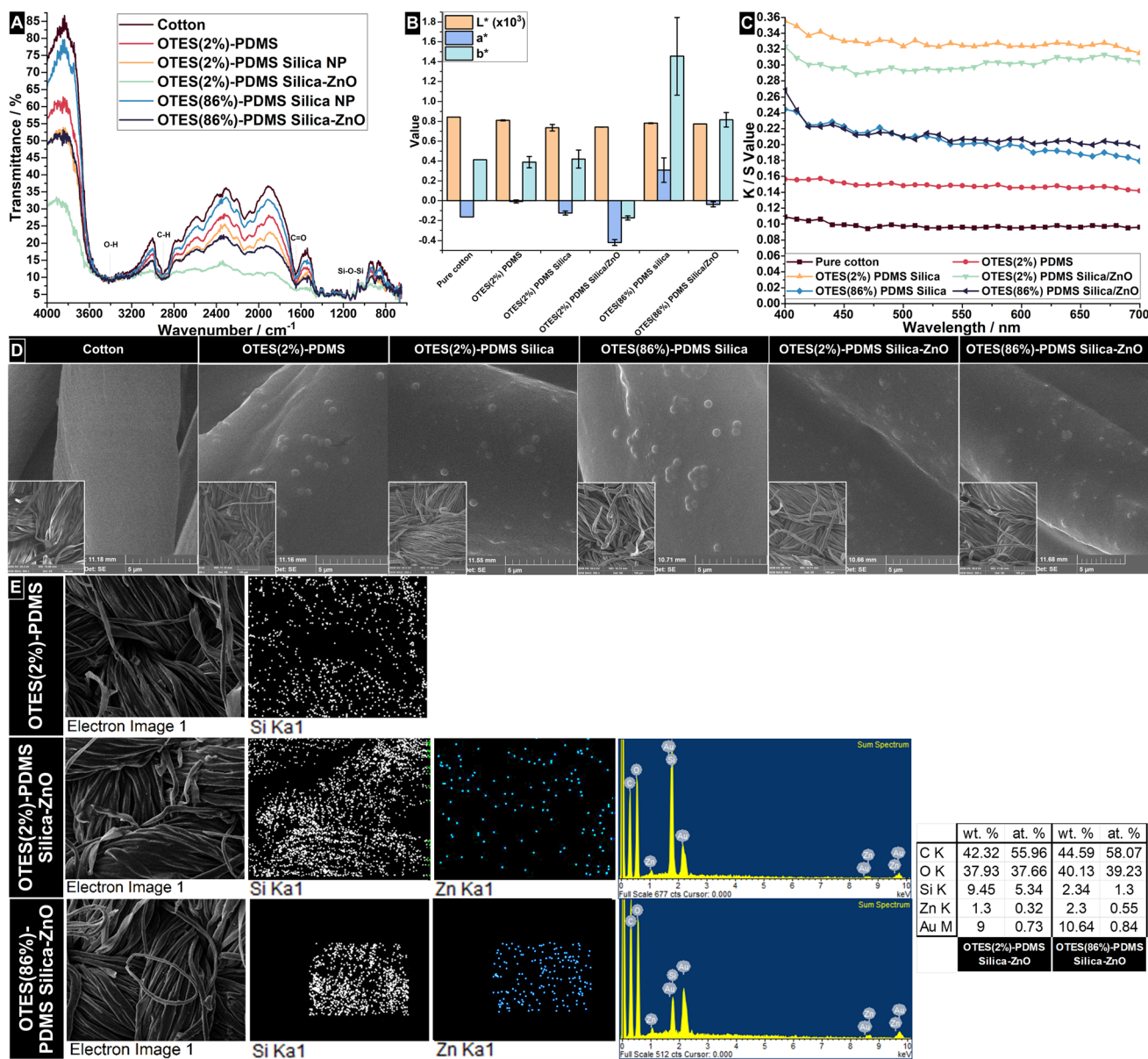


Fig. 3 (A) FTIR transmittance spectra; (B)  $L^*a^*b^*$  values; (C) K/S spectra; (D) top-down SEM images (scale 5  $\mu\text{m}$  main image; scale 100  $\mu\text{m}$  inset), of coated cotton substrates, and; (E) EDS elemental analyses of coated fabric substrates comparing Zn-loading.

and increased the persistence in hydrophobicity, *i.e.*, stable WCA values over 20 s; a critical factor for water-repellent wearables. The variant data between silica and ZnO NP may be explained by the different mechanisms of interaction between the different NP; the less effectively hydrophobic ZnO being more present with OTES results in lower water-repellency values.

NP-addition also afforded the greatest improvements in roll-off; the lowest sliding angle was achieved with silica-ZnO ( $12^\circ$ ), indicating improved mobility of water droplets on the surface. The varied topography caused by the two different NP coating types on the fabric surface may be related to the changes in WCA although the rough, fibrous structure of textile-based surfaces may intrinsically increase the sliding angle.<sup>97</sup> The NP seemingly serve to decrease the tilt angle although no sample reaches the threshold ideal of  $<10^\circ$ .

UV-induced superhydrophilicity is a long-reported phenomenon; short bouts of UV-irradiation on surfaces containing, *e.g.*, photoactive ZnO, allowing for remarkable changes of existing surface wettability properties (*e.g.*, superhydrophilic to superhydrophobic, or *vice versa*).<sup>98</sup> However, in this study, whenever ZnO was used, no meaningful changes in wettability were observed, *i.e.*, even after 40 h of 370 nm UV light exposure, which suggests that silica-ZnO combinations broadly stabilize ZnO to light-induced changes – an extremely valuable steadying mechanism for reliable and consistent performance of coatings over an extended period.<sup>99–101</sup> While UV-curing of the PDMS and/or OTES may be a contributory factor, the fact that marked improvements were not seen in samples absent NP addition indicates that it has a minor effect.<sup>102,103</sup>



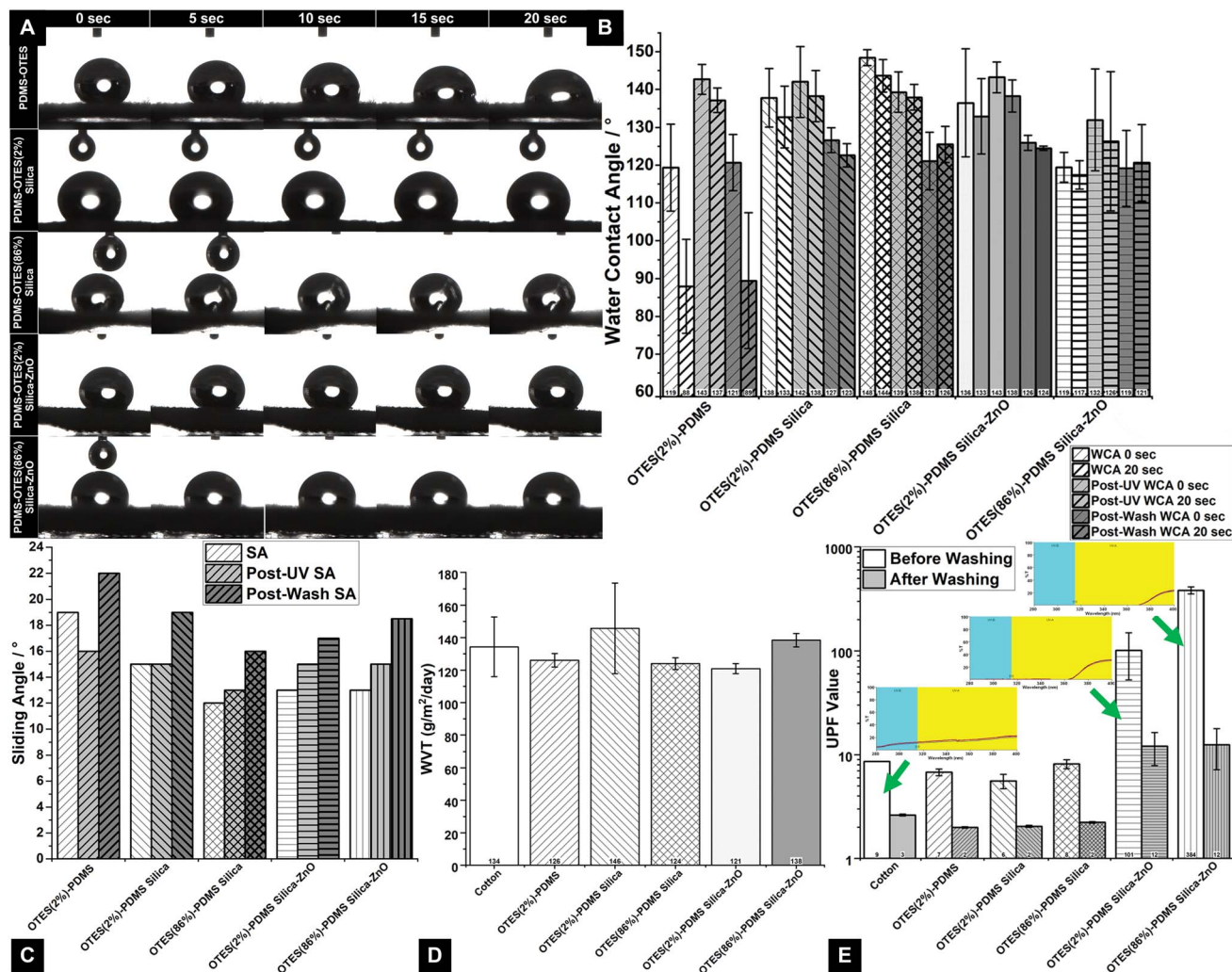


Fig. 4 Coated fabric substrate comparisons; (A) illustrative water contact angles (WCA) of the relative persistence of repellency over time for as-prepared samples; (B) average WCA for; as-obtained samples; after 40 h exposure to 370 nm, or; 3 accelerated laundering cycles; (C) sliding angle values for; as-obtained samples; after 40 h exposure to 370 nm, or; 3 accelerated laundering cycles; (D) water vapor transmission rate values of as-obtained samples, and; (E) fabric UV-protective factor pre- and post-laundering (illustrative ultraviolet-visible experimental transmittance spectra of as-prepared samples, inset).

The WVTR analyzes how much vapor passes through a material for a given size, under constant temperature and humidity conditions. Coated fabrics retained similar water vapor permeability properties, with minor decreases of  $\sim 8\%$ , as compared to the uncoated cotton fabric references (Fig. 4d), which are acceptable for real-world usage. Thus, the hydrophobizing agent and the NP coatings had minimal effects on cotton fabric pore blockage, likely due to the relatively sparse, discontinuous coatings that result from the single-coating approach adopted, which is ideal for thermal comfort in wearables applications.<sup>53,104</sup> In cases where the treated sample's performance is seemingly superior to the pure cotton fabric, this is seen as an experimental variation effect. Thus, the great enhancements in both water repellence and UV protection were gained without any significant change in water vapor permeability. An extremely positive outcome.

UPF values for pure cotton and treated fabrics show that addition of silica-ZnO NP has offers at least a one order of magnitude improvement on UV-blocking capability, but only where ZnO is present (Fig. 4e).<sup>105</sup> This includes near-complete suppression of UV light transmission through the coated substrates across 280–370 nm (*i.e.*, all of UV-B and much of the UV-A) – the higher the ZnO loading, the better the UV-shielding improvement. A higher amount of OTES usage also seems to allow for greater ZnO inclusion, due to its superior grafting capability as compared to PDMS.<sup>106</sup> Samples were also found to be UV-irradiation resistant – there was an essentially unchanged UPF property before and after UV (370 nm) 40 h irradiation, within experimental error. Without the surface modification contributions of the ZnO additive specifically, the UPF values were unremarkable and indistinguishable from the uncoated reference substrate.



Impermanence to washing is frequently encountered for fabric coatings. However, silica has often been reported as an effective method for stabilizing nanomaterials on textiles.<sup>107</sup> Empirically, UPF, water contact angle and sliding angle data were reacquired for all samples after three accelerated wash cycles of laundering (*i.e.*, equivalent to nine standard domestic wash cycles). While performance deteriorations were seen in all conditions, the performance retention in coated fabrics as compared to the cotton reference standard showed that at least some coating material remained adhered to the substrate. Thus, despite some loss of coating material, there is a degree of persistence in the silica-ZnO NP functional performance, such that the substrate could still be reused for water-repellency and/or UV-protective functions. It should be noted that treatment of any kind will have a deleterious effect on pristine cotton fabric, and this explains the discrepancy in values between the pristine substrates, and the chemically-treated, coated variants, even without the laundering steps.

## 4. Conclusions

This work provides sustainable routes to functionally-coated fabrics with UV-shielding capability and stable, high hydrophobicity (up to 148°), while still retaining excellent water vapor permeability properties. Woven cotton substrates were alternatively dip-coated with either silica or silica-ZnO NP. The resultant coatings were sparsely distributed throughout the entire fibrous surface. The choice of binder/hydrophobizing agent was important in decreasing the surface energy, increasing surface roughening and NP-retention; a higher relative OTES binder concentration allowed for greater additive loading onto the substrate.

These NP intermediate coating layers provide otherwise rough, porous fabric substrates with stable and reliably high hydrophobic properties with low sliding angles. Fabrics imparted with silica-ZnO NP UV-absorbers exhibited markedly superior UV shielding properties with UPF values far in excess of 50, albeit with a slight decrease in relative hydrophobic character as compared to the solely silica-based coatings. Such combined UPF and water contact angle properties were stable even after 40 h continuous exposure to UV-radiation, avoiding the otherwise commonly encountered UV-induced hydrophobic-to-hydrophilic transition for ZnO. These properties also showed persistence, albeit with an associated performance decline, after being subjected to accelerated laundering treatments. Thicker coatings and/or more resilient binders could perhaps overcome this issue, for longer-lasting performance retention.

Durably hydrophobic and UV protective textiles have potential applications in outdoor garments and architectural fabrics, which could help defend against extreme weather environments, protecting the wearer during open-air activities. However, an ideal treatment for scalability would involve water-processable polymers and methods, and so further procedural refinement in this direction would be valuable. In addition, further study is necessary to characterize the mechanisms contributing to these hydrophobic surfaces. Such work is underway.

## Consent for publication

All authors agree to the submission and publication of article materials.

## Author contributions

All authors contributed meaningfully to all aspects of experimentation, analysis and composition.

## Conflicts of interest

There are no known conflicts to declare.

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