


 Cite this: *RSC Adv.*, 2024, 14, 46

An investigation of structural, thermal, and electrical conductivity properties for understanding transport mechanisms of CuWO_4 and $\alpha\text{-CuMoO}_4$ compounds

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Recently, inorganic oxide components with high ionic conductivity have been widely explored due to their high stability, safety, and energy density properties. In this context, the present work focuses on the inorganic oxides CuMO_4 ($M = \text{W, Mo}$), which have been successfully synthesized using the traditional solid-state method. They were characterized by X-ray powder diffraction, thermal analysis, and complex impedance spectroscopy. X-ray diffraction data refined *via* the Rietveld method indicated that these compounds are well crystallized in the triclinic system with the $P\bar{1}$ space group. Besides, the electrical conductivity behavior of these materials was analyzed using the impedance spectroscopy technique in the frequency range of 10^0 to 10^6 Hz and in the temperature range of 443 K to 563 K. The absence of a phase transition observed in the calorimetric study was confirmed by the σ_g and ω_h variations as a function of temperature. The AC conductivity was analyzed by Jonscher's power law. The outcomes of the study on charge transportation in CuMO_4 (where $M = \text{W, Mo}$) suggest that the overlapping large polaron tunneling (OLPT) mechanism was present in CuMoO_4 , while the correlated barrier hopping (CBH) and the non-overlapping small polaron tunneling (NSPT) were present in CuWO_4 . A correlation between the crystal structure and the ionic conductivity was established and discussed. For the two title compounds, modulus analysis revealed that the charge carriers were mobile over short and long distances at low and high frequencies, respectively. The temperature variation of the M' peak showed a thermally activated relaxation process.

Received 2nd November 2023

Accepted 3rd December 2023

DOI: 10.1039/d3ra07453f

rsc.li/rsc-advances

1. Introduction

During the past decade, interest in the relationship between crystal structures and physical properties has experienced a resurgence due to the discovery of high-temperature superconductivity and its dependence on details of the crystal structure. There are various typical structure types for binary oxides with the general formula AMO_4 , in which A and M cations can be multivalently and compositely replaced.

Many AMO_4 oxides, such as arsenates, chromates, silicates, vanadates, and phosphates, are isostructural with zircon ($I4_1/a$, No. 141, $Z = 4$); a few AMO_4 oxides, such as periodates, germinates, tungstates, and molybdates, crystallize in the scheelite structure ($I4_1/a$, No. 88, $Z = 4$); a large number of

tantalates, molybdates, and tungstates crystallize with the wolframite structure ($P2_1/c$, No. 13, $Z = 2$); relatively few AMO_4 oxides exist as M' -fergusonite structures ($P2_1/c$, No. 14, $Z = 2$) or M' -fergusonite under normal conditions, but lots of AMO_4 oxides transform from zircon or scheelite structures to M' -fergusonite and M' -fergusonite under pressure.

Among these, the wolframite structure AMO_4 (where A denotes transition metal ions and M denotes divalent cations) is particularly notable. Copper oxides (CuMO_4 , with $M = \text{W, Mo}$) are important inorganic materials that have gained enormous scientific importance due to their wide range of applications, including optical fibers, scintillator materials, industrial catalysts, microwave applications, humidity sensors, and photoluminescent materials, as well as their magnetic, electric, and electrochemical properties.¹⁻⁷ Among the molybdates, the CuMoO_4 compound exhibits highly complex polymorphism.⁸⁻¹¹ To date, six different polymorphs of CuMoO_4 have been identified in the literature: namely ambient condition $\alpha\text{-CuMoO}_4$ (green phase),⁸ high-temperature $\beta\text{-CuMoO}_4$,⁹ low-temperature $\gamma\text{-CuMoO}_4$ (brownish red phase),¹⁰ high pressure (HP) $\text{CuMoO}_4\text{-II}$,¹¹ distorted wolframite $\text{CuMoO}_4\text{-III}$,¹² and monoclinic $\varepsilon\text{-CuMoO}_4$.¹³ Both the compounds $\text{CoMoO}_4\text{-II}$ and

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CuMoO-III are isostructural to the tungstates CoWO₄ and CuWO₄, respectively.¹⁴

Additionally, copper tungstate (CuWO₄) is regarded as a superior material for creating the positive electrodes of rechargeable Li batteries¹⁵ and for synthesizing W-Cu pseudo alloys with a copper content of 10–35 mol%, which are almost free of pores.¹⁵

However, there are still a lot of unanswered questions regarding the physical properties of CuMO₄ (M = W, Mo).

A comprehensive examination of the literature revealed that there have been no reports on the electrical properties of α -CuMoO₄ and CuWO₄. In order to correlate these characteristics to the structural aspects of these materials, we consequently decided to record the electrical and dielectric properties as a function of frequency and temperature. Additionally, this study could provide important information about the conduction process and the electrical conductivity of the material.

2. Experimental methods

2.1. Synthesis procedure

In a standard process, polycrystalline CuMO₄ (M = W, Mo) powder samples were synthesized *via* the conventional solid-state reaction by mixing high-purity precursors of CuO (Sigma-Aldrich, $\geq 99\%$), MoO₃ (Sigma-Aldrich, $\geq 99\%$), and WO₃ (Sigma-Aldrich, $\geq 99\%$).

In the first step, the starting precursors were mixed in stoichiometric quantities and meticulously ground in an agate mortar initially, sealed in platinum crucibles, and then heated at 773 K for 16 h. After heat treatment, the samples were ground up and then reheated again between 923 K and 1023 K to ensure homogeneity, as summarized in Table 1. Afterward, the reaction products were cooled at a rate of 393 K h⁻¹ to room temperature.

Finally, the colors of these synthetic samples, which varied depending on the metal used, were in accordance with those revealed in the literature, as shown in Table 1.

Note that the following stages are sequentially reached with increasing sintering temperature: the formation of isthmuses between the grains, a compaction of grains, and finally, their growth.

The crystallization and purity of these phases were determined by X-ray diffraction (XRD) of the powder. Then, based on the Rietveld method, the data collected at room temperature were analyzed by the Fullprof software. Differential scanning calorimetric analysis was carried out using a DSC (Q-100) TA instrument with a scanning rate of 10 °C min⁻¹ between 283 K and 483 K in temperature. To carry out the electrical measurements, the powder resulting from the grinding process was

shaped into a pellet (8 mm in diameter and 1.1 mm in thickness) in a uniaxial hydraulic press at a pressure of 3 tons per cm². The measurements were performed using a 1260 Solartron Impedance Analyzer under vacuum, operating in the temperature range of 443 K to 563 K.

3. Results and discussion

3.1. Crystalline parameters

The purity of the synthesized compounds and their crystalline characteristics were determined through XRD patterns recorded at room temperature. Fig. 1(a) and (b) illustrates the experimental and calculated XRD profiles of the compounds α -CuMoO₄ and CuWO₄, respectively, as well as the corresponding discrepancies. The full pattern refinement was analyzed by the Rietveld method using the Fullprof program. The lattice parameters and the $P\bar{1}$ space group reported in the bibliographic data were used as the starting point.^{8,16} All the Bragg peaks were successfully indexed and satisfactorily modeled. The quality factors that show good agreement between the experimental (red dots) and calculated (black solid line) profiles are $\chi = 4.02$ and $\chi = 3.1$ of the compounds α -CuMoO₄ and CuWO₄, respectively. Indeed, we could notice the presence of a small extra peak identified by an asterisk in the X-ray powder pattern (Fig. 1a) of α -CuMoO₄, indicating the presence of a secondary phase of MoO₃ in the vicinity of 2%, which crystallizes in the orthorhombic system.¹⁷ The unit cell parameters of MoO₃ are reported by Nora Wooster (2015).

The estimation results of the structural parameters are illustrated in Table 2. These results are in good agreement with the literature.^{8,16} This study demonstrated that the system and crystal structure could be changed by switching the divalent cation (from W²⁺ to Mo²⁺).

3.2. Structural description

The structure of CuWO₄ is implied in Fig. 2a. In this structure, the copper atom is surrounded by six oxygen atoms, four of which have an approximately square planar configuration and the remaining two have a longer bond distance, providing an elongated octahedron. The tungsten atom is located within a slightly distorted octahedron but is considerably displaced from its center.

The WO₆ and CuO₆ octahedra are connected by their corners, and the CuO₆ octahedra exhibit an elongated pseudotetragonal geometry due to the Jahn–Teller effect of the Cu²⁺ cation.^{16,18} It is clear from the figure that although CuWO₄ is triclinic, its structure is topologically related to that of monoclinic wolframite ($P2_1/c$). The W–O distances of the WO₆ octahedra are slightly distorted, similar to those of wolframite, and range from 1.79 Å to 2.20 Å.

Nevertheless, the monoclinic $P2_1/c$ symmetry is lowered to the triclinic $P\bar{1}$ symmetry due to the Cu²⁺ Jahn–Teller effect, which decreases the degeneracy of the 3D orbitals.¹⁸

As a result, the CuO₆ octahedra adopt an elongated pseudotetragonal geometry with two axial Cu–O distances near 2.50 Å and four planar Cu–O distances at 1.93 Å.

Table 1 Conditions of the synthesis of CuWO₄ and α -CuMoO₄

Sample	Precursors	Synthesis	Color
CuWO ₄	CuO, WO ₃	773 K/16 h, 1023 K/10 h	Light yellow
α -CuMoO ₄	CuO, MoO ₃	773 K/16 h, 923 K/14 h	Green



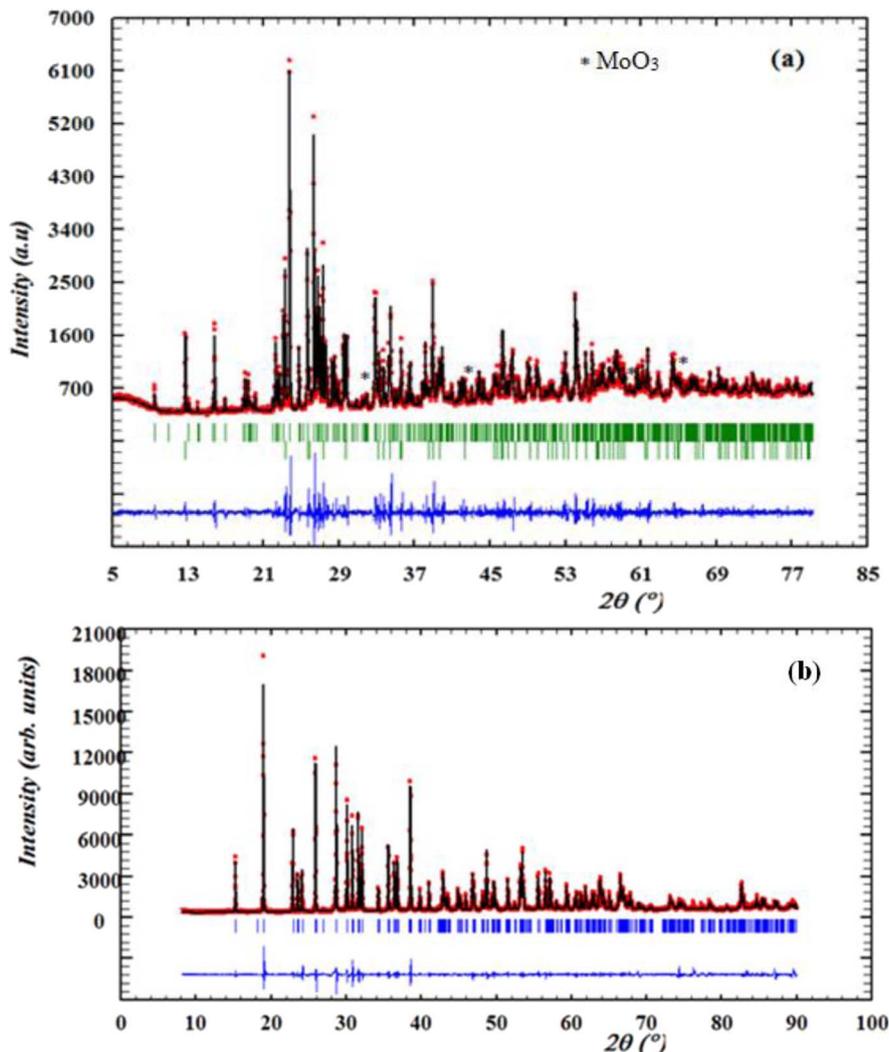


Fig. 1 X-ray diffraction patterns and Rietveld refinement of α -CuMoO₄ (a) and CuWO₄ (b) samples recorded at room temperature.

Additionally, the alternating CuO₆ and WO₆ octahedra form endless zigzag chains by sharing the edges of copper and tungsten atoms in the CuWO₄ sequence of layers that are located between the oxygen sheets. In contrast, as illustrated in Fig. 2b, the α -CuMoO₄ structure can be characterized by MoO₄

tetrahedra, CuO₅ square pyramidal polyhedra, and CuO₆ distorted octahedra, which are joined by common corners and edges to create layers. For octahedrally coordinated CuO₆, the average Cu–O distances in the α -CuMoO₄ range from 1.94 Å to 2.53 Å. However, for pyramidal CuO₅ (C_{4v}) they vary from 1.90 Å

Table 2 Crystallographic data of the structure refinement for CuWO₄ and α -CuMoO₄

Formula	α -CuMoO ₄	CuWO ₄
Unit lattice parameters (Å)	$a = 6.789229$ (5) $b = 8.37269$ (3) $c = 9.906073$ (4) $\alpha = 96.892^\circ$, $\beta = 107.009^\circ$, $\gamma = 101.132^\circ$ $V = 568.70$ (2)	$a = 4.7060$ (4) $b = 5.8415$ (5) $c = 4.8817$ (4) $\alpha = 91.881^\circ$, $\beta = 92.497^\circ$, $\gamma = 82.785^\circ$ $V = 136.89$ (3)
Unit cell volume (Å ³)		
Crystal system	Triclinic	Triclinic
Space group	$P\bar{1}$	$P\bar{1}$
R_{exp} (%)	10.66	06.2
R_p (%)	22.00	1.77
R_{wp} (%)	23.08	1.53
χ^2	4.02	3.1



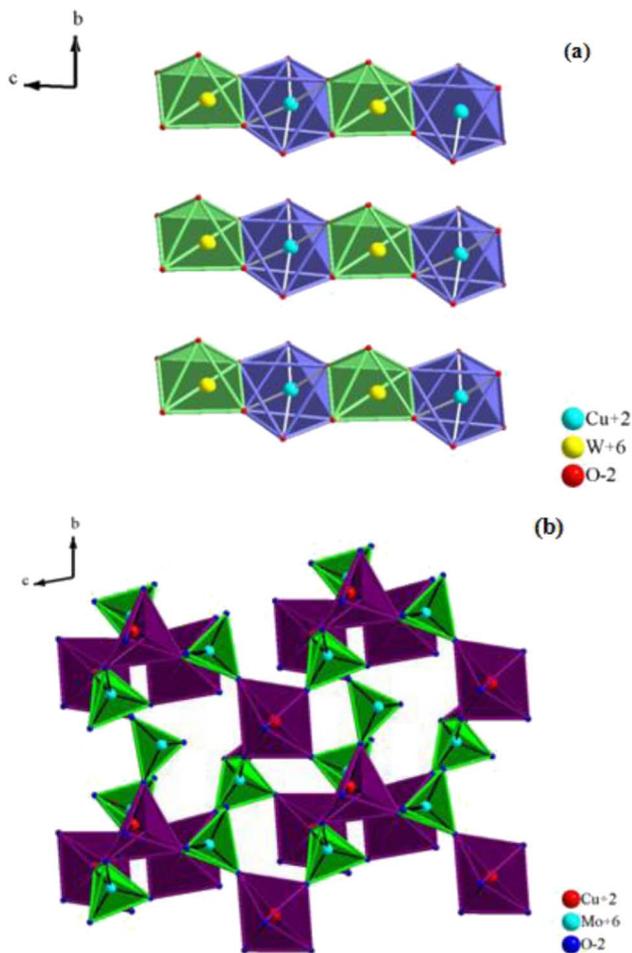


Fig. 2 The crystal structure of CuWO_4 (a) and $\alpha\text{-CuMoO}_4$ (b) along the a axis.

to 2.53 Å, while the Mo–O bond distances range from 1.74 Å to 2.38 Å.

It is important to note that the symmetry elements and the relative positions of the atoms distinguish the difference between the structures of CuWO_4 and $\alpha\text{-CuMoO}_4$. This appears to be caused, most probably, by the potent Jahn–Teller effect present in CuO_6 octahedra, which causes a significant elongation. Also, the geometry of the MO_6 ($M = \text{W}, \text{Mo}$) and CuO_6 octahedra and their connection have a significant impact on the band gap as well as the associated optical, electrical, and magnetic properties.

3.3. Thermal analysis

Differential scanning calorimetry is typically used to examine the thermal transformation of a material, such as crystallization, transition to glass, melting, *etc.* To prevent the material from reacting with the atmosphere, this analysis was carried out under a supply of an inert gas (nitrogen). The results of the thermal analysis (DSC) for the CuMO_4 ($M = \text{W}, \text{Mo}$) samples in the heating temperature range of 283 K to 483 K and with a ramp rate of 10 °C min^{-1} are displayed in Fig. 3. We notice that this study shows that no exothermic or endothermic peaks

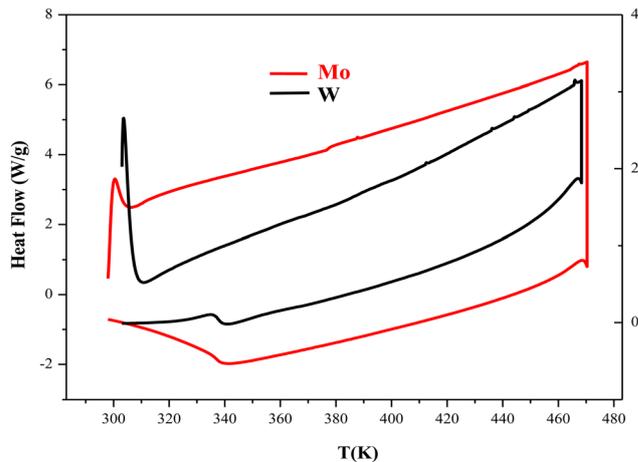


Fig. 3 DSC thermograms of $\alpha\text{-CuMoO}_4$ and CuWO_4 compounds.

are displayed in the temperature range studied, which can be attributed to the absence of structural transitions in the two compounds.

3.4. Electrical properties

3.4.1. The Nyquist diagram and equivalent circuit. After the structural and thermal analysis, the electrical characteristics of the compounds were carried out by complex impedance spectroscopy (CIS), which operates over a wide range of temperature and frequency. It provides information on the microstructure of various materials, including the electrode interfaces, grains, and grain boundaries. This versatile technique gives information on the mobility of individual components within the system and the occurrence of specific interactions. It displays a direct connection between the response of the real system and the ideal circuit made by the electrical components. Through this method, a specimen undergoes a low sinusoidal perturbation. The impedance is evaluated by analyzing the current response to the varying frequency of the imposed alternating voltage over a wide range, and subsequently, it is separated into real and imaginary components.

Indeed, the complex impedance Z^* is divided into its real part Z' and its imaginary part Z'' :

$$Z^* = Z'(w) - jZ''(w) \quad (1)$$

Fig. 4(a), (b) and 5(a), (b) depict the Cole–Cole plot of the typical complex impedance plane for CuMO_4 ($M = \text{W}, \text{Mo}$) samples at a temperature interval ranging from 443 K to 563 K. It is worth noting that the centers of the depressed semicircles are located below the real axis, indicating that both materials exhibit non-Debye behavior.¹⁹

For the Mo-based compound, the experimental data showed the existence of two semi-circles, each placed within a different frequency domain, indicating the presence of two contributions. The high-frequencies semicircle is attributed to the grain effect and the other at low frequencies is attributed to the effect



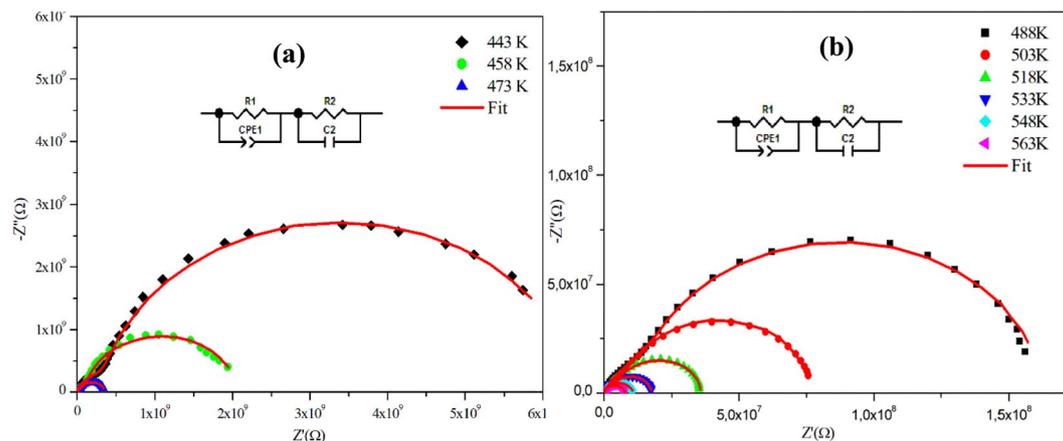


Fig. 4 (a) and (b) Complex impedance spectra in the Nyquist plane with an electrical equivalent circuit (inset), accompanied by theoretical data (solid line) for α -CuMoO₄.

of the grain boundary.²⁰ On the other hand, to visualize these two contributions (grains and grain boundary elements), we performed a deconvolution of the Nyquist curve recorded at $T = 563$ K (Fig. 6).

Actually, we can distinguish between the two contributions by looking at the resistance value provided by the adjustment; the arc with the lowest resistance value is attributed to the effect of the grain, whereas the other is attributed to the grain boundary effect. The higher resistivity of the grain boundary is reflected in the increased arc dispersion at low frequencies.

Based on the ZView software,²¹ the $-Z'' = f(Z')$ plots were adjusted by an equivalent electrical circuit formed by a resistance (R_g) parallel to the fractal capacitance (CPE_g) describing the response of the grains in series with a resistance (R_{gb}) parallel to the capacitance (C_{gb}) describing the grain boundary (inset of Fig. 4).

Whereas, for the W-based compound, these spectra are composed of a single semi-circular arc at each temperature suggesting the dominance of the grain contribution.

The spectra were adequately modeled by a single-cell circuit model, formed by a resistance (R_g) in parallel with the series combination (C_g and CPE_g) as depicted in the inset of Fig. 5.

In this model, R_g introduces the resistance, C_g represents the capacitance, and CPE_g is the non-ideal capacitor usually known as the constant phase element.

Furthermore, we observe that according to Fig. 4 and 5, the radii of the semicircles decrease with increasing temperature in the Cole-Cole plot, suggesting that the conduction mechanism is thermally activated. The latter is known as the negative temperature coefficient of resistance (NTCR), which is a characteristic of semiconductor compounds.²²

Indeed, it is important to note that the non-ideal Debye behavior and the observed depression of the semicircles are both explained by the presence of a constant phase element (CPE) in these equivalent circuit models.²³

The impedance of the CPE contribution is an empirical function of the following type:

$$Z_{\text{cpe}} = \frac{1}{Q \times (j\omega)^\alpha} \quad (2)$$

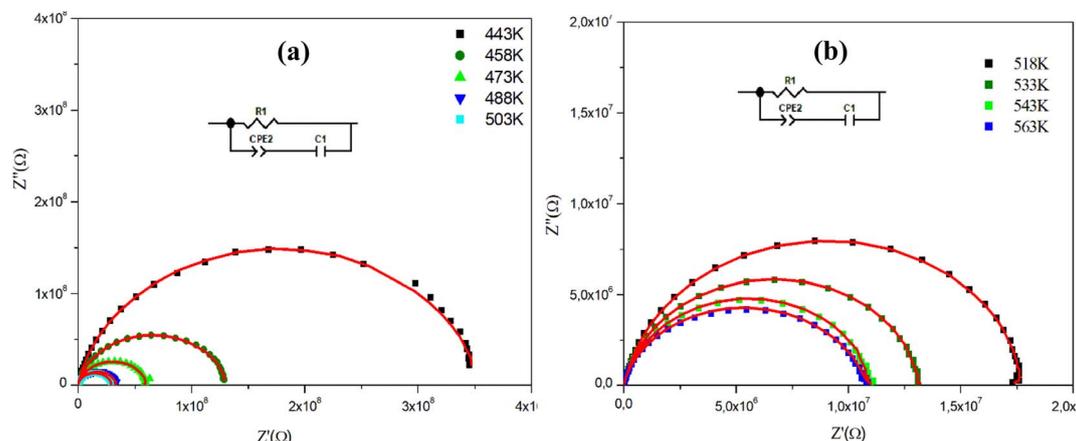


Fig. 5 (a) and (b) Nyquist diagram spectra as a function of temperature and the electrical equivalent circuit (inset) of the CuWO₄ compound.



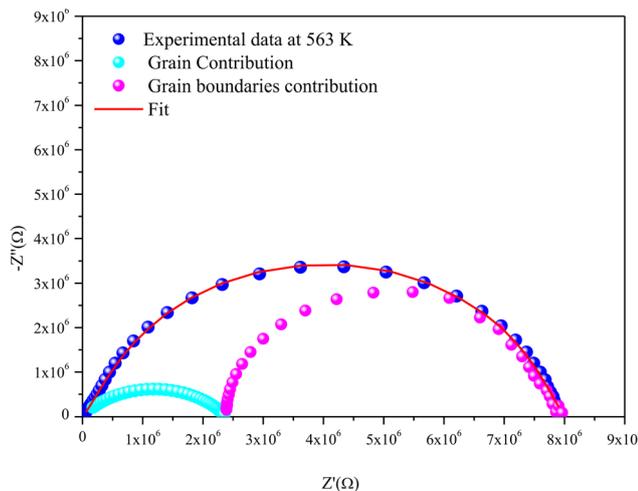


Fig. 6 Deconvolution of the Nyquist curve recorded at $T = 563$ K.

where Q is the capacitance value of the CPE element; ω is the angular frequency; and α is a parameter indicating the change of the compressed semicircle from an ideal semicircle.

The frequency behavior of the complex impedance parts ($-Z''$) of the α -CuMoO₄ and CuWO₄ samples, for distinct temperatures, is displayed in Fig. 7(a), (b) and 8(a), (b), respectively.

The spectra explicitly describe the strength and the position of the relaxation peak which is found to be different for several sintering temperatures. We can observe that the impedance plots are characterized by the presence of maximum peaks (Z''_{\max}) at a specific frequency known as the relaxation frequency, f_{\max} . The relaxation peak shifts to very high frequencies as the temperature rises. These findings demonstrate the presence of the electrical relaxation phenomenon, characterize its nature, and highlight its strength in our material.^{24,25} As temperatures increase, a significant broadening of the asymmetric peaks is observed, indicating a temperature-dependent relaxation process in the system that lengthens the relaxation period (as depicted by the increased width of the

peaks). This may be explained by the fact that the relaxation process is caused by defects or vacancies at high temperatures, while immobile species or electrons are accountable at low temperatures.^{26,27}

Additionally, the height of the relaxation peaks is also impacted in addition to peak broadening. As the temperature rises, the height of the relaxation peaks declines, suggesting a loss of resistive characteristics. It also suggests the existence of a relaxing process that is initiated by heat and implies that the relaxation mechanism is thermally stimulated.

Finally, for all temperatures and for higher frequencies, the imaginary part of $-Z''$ is merged, which probably indicates an accumulation of space charges in the system. There was also a perfect superposition between the experimental scatter data and the theoretical line for these curves.

The advantageous justification for the choice of the equivalent circuit retained is verified by the variations of the experimental values of (Z') and ($-Z''$) according to those calculated using the parameters of the equivalent circuit model at different temperatures (Fig. 9(a), (b) and 10(a), (b)). It is clear from these graphs that the slope produced by a linear fitting of the data points at all temperatures is almost equal to the unit. This behavior leads us to the conclusion that the chosen equivalent circuit accurately describes the electric characteristics of the samples and that there is a good correlation between the theoretical (calculated line) and experimental data (scatter) of the real and imaginary impedance parts.

3.5. Conductivity studies

3.5.1. DC conductivity. The conduction process in amorphous semiconductors involves temperature-dependent Dc conductivity. Electrical conductivity is a thermally activated process that relies on the ordered movement of lightly bound charged particles while being affected by a continuous electric field. The following equation can be used to determine the electrical conductivity of the grain (σ_g) at each temperature, derived from the resistance values obtained from the equivalent circuit:²⁸

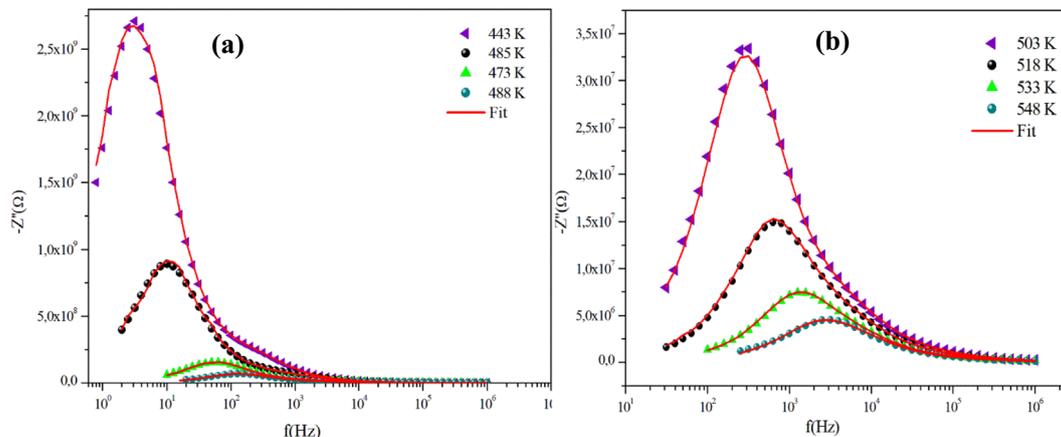


Fig. 7 (a) and (b) Variations of the experimental and calculated imaginary parts of impedance ($-Z''$) as a function of frequency at some representative temperatures for α -CuMoO₄.



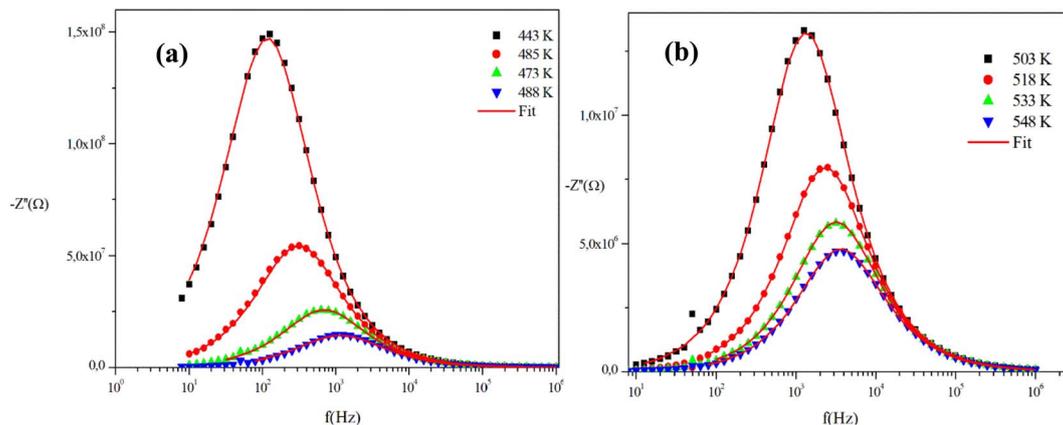


Fig. 8 (a) and (b) Variations of the experimental and calculated imaginary parts of impedance ($-Z''$) as a function of frequency at several temperatures for the CuWO_4 compound.

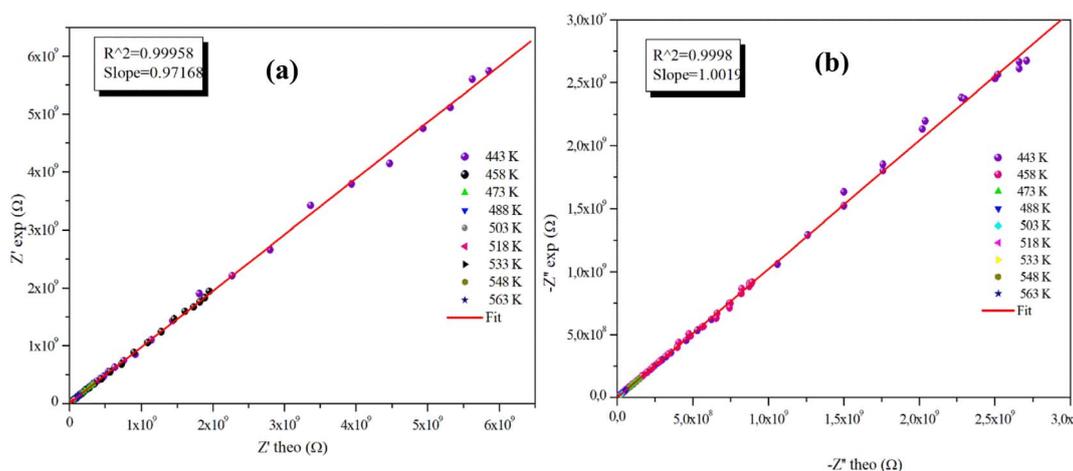


Fig. 9 (a) and (b) Plots of measured values versus simulated values of the real and imaginary parts of the impedance for the $\alpha\text{-CuMoO}_4$ compound.

$$\sigma_g = \frac{e}{R_g S} \quad (3)$$

where R_g is the resistance of grain determined by the adjustment, S is the area of the electrode that is deposited on the pellet, and e is the thickness of the sample.

Fig. 11(a) and (b) displays the conductivity (σ_g) of the $\alpha\text{-CuMoO}_4$ and CuWO_4 samples in a temperature range of 443 K to 563 K, at a constant voltage of 1.5 V. These curves clearly demonstrate that the response of $\ln(\sigma_g T)$ versus $1000/T$ represents a straight line or a continuity of the curve slope, indicating the absence of a phase transition in these samples over the temperature range studied, which was verified by the DSC diagram (Fig. 3).

Indeed, as can be seen from the graph, the Dc conductivity rises linearly over the entire temperature range as a result of an increase in the number of charge carriers that are made available by the increase in thermal energy. This is due to the thermally assisted tunneling of charge carriers in

the band tail of the localized,²⁹ which leads to the Arrhenius equation:

$$\sigma_g T = A \exp\left(\frac{-E_{a,g}}{K_B T}\right) \quad (4)$$

where K is the Boltzmann constant, A is the pre-exponential factor, which includes the mobility of the charge carrier and the density of states, T is the absolute temperature, and $E_{a,g}$ is the activation energy for the conductivity of grain. The values of the activation energies obtained from the linear fit of the data points are $E_{a,g} = 0.73$ eV for CuWO_4 , and $E_{a,g} = 0.60$ eV for $\alpha\text{-CuMoO}_4$.

Table 3 presents a comparison between the electrical properties of CuWO_4 and $\alpha\text{-CuMoO}_4$. The table shows that $\alpha\text{-CuMoO}_4$ has a higher conductivity due to its lower activation energy $E_{a,g}$. The influence of structural features on the conductivity and activation energy of each phase can be deduced from these data. For the $\alpha\text{-CuMoO}_4$ compound, the highest conductivities and the lowest $E_{a,g}$ are correlated with the better mobility of the Mo cation in the corresponding frameworks. In fact, the greater Mo–O bond distances (1.74 Å–2.38 Å)



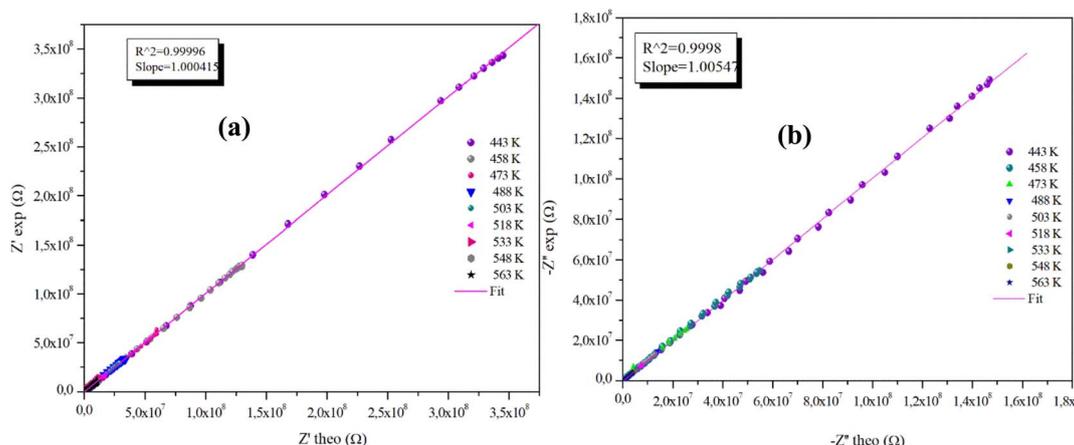


Fig. 10 (a) and (b) Plots of measured values versus simulated values of the real and imaginary parts of the impedance for the CuWO_4 compound.

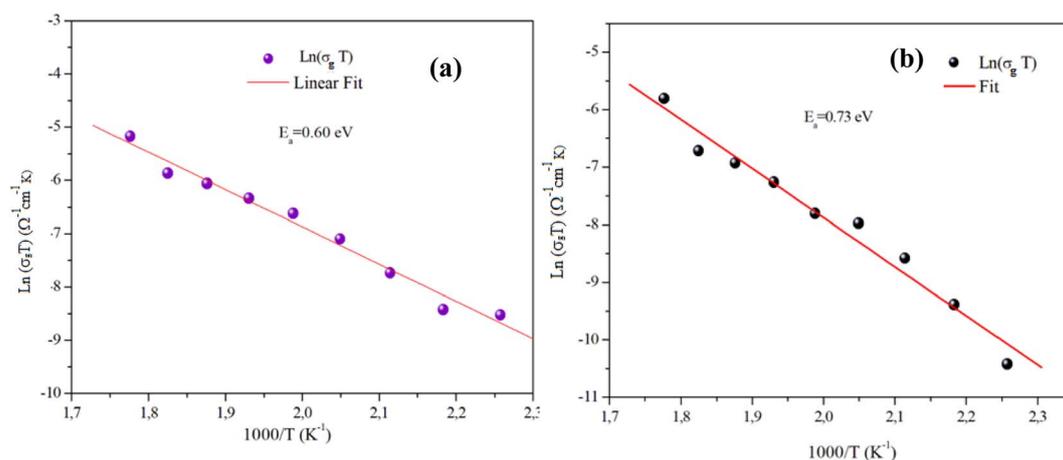


Fig. 11 Arrhenius plots for the bulk conductivity of the (a) $\alpha\text{-CuMoO}_4$ and (b) CuWO_4 samples.

Table 3 Activation energies and conductivities at 518 K for CuMO_4 ($M = \text{W}, \text{Mo}$) samples

Formula	$\alpha\text{-CuMoO}_4$	CuWO_4
$E_{a,g}$ (eV)	0.60	0.73
W–O (\AA)	—	1.79–2.20
Mo–O (\AA)	1.74–2.38	—
$\sigma_{503\text{ K}}$ ($\Omega^{-1}\text{ cm}^{-1}$)	2.6474310^{-6}	8.10010^{-7}
A ($\Omega^{-1}\text{ cm}^{-1}\text{ K}$)	2627.474	2262.84

compared to those of W–O (1.79 \AA –2.20 \AA) result in a weakening of the attraction between Mo^+ and O_2 and a release of the ion attachment to the crystal, which lowers the potential barrier and requires a lower activation energy, thereby increasing conductivity. In addition, it is also associated with the dimension of the tunnels in $\alpha\text{-CuMoO}_4$.

On the other hand, it appears that the conductivity of $\alpha\text{-CuMoO}_4$ is slightly higher than that of the CuWO_4 compound, which is due to the smaller mass and ionic radius of molybdenum ($m_{\text{Mo}} = 1.59310^{-22}$ g, $r_{\text{Mo}^+} = 0.59$ \AA) compared to

tungsten ($m_{\text{W}} = 3.052 \cdot 10^{-22}$ g, $r_{\text{W}^+} = 0.60$ \AA) and consequently it is more mobile than W^+ .

Additionally, the large value of the pre-exponential factor, which is proportional to the density of the charge carriers and the disorder within the material, can also be used to explain the increase in conductivity of the compound of $\alpha\text{-CuMoO}_4$.

Indeed, the pre-exponential factor obtained thanks to the Arrhenius parameterization law is given by:³⁰

$$A_0 = \left(e^2 \times a_h^2 \times \frac{v_0}{6K_B} \right) \times N(T) \times \exp\left(\frac{S_\mu}{K_B}\right) \quad (5)$$

where a_h is the jump distance, v_0 is the natural frequency of vibration, $N(T)$ is the number of charge carriers, and S_μ is the migration entropy.

3.5.2. AC electrical conductivity. The nature of defect centers in disordered systems has been extensively studied using AC conductivity measurements, which should account for this type of conduction. In most polar semiconductor oxides, the high-frequency dependence of the AC conductivity is often observed in experimental settings. This observation can then be explained in terms of thermally stimulated charge carrier



hopping between sites localized above the potential barrier. In fact, an empirical formula is used to determine the conductivity values of the material (σ_{ac}):

$$\sigma_{ac}(\omega) = \frac{e}{S} \times \frac{Z'}{Z'^2 + Z''^2} \quad (6)$$

where e/S is the geometric report of the pellet and Z' and Z'' are the real and the imaginary parts of the complex impedance data.

The variation of the AC conductivity of CuMO_4 ($M = \text{Mo}, \text{W}$) samples at different temperatures is shown in Fig. 12(a) and (b). These spectra are described by two regions and are theoretically analyzed using Jonscher's universal power law:^{31,32}

$$\sigma_{ac}(\omega) = \sigma_{dc} + A\omega^s \quad (7)$$

where σ_{dc} represents the value of the conductivity of the direct current at low frequency; A is a parameter dependent on temperature that determines the strength of polarizability; and s is the power law exponent that reflects the extent to which the charge carriers interact with the lattice of CuMO_4 ($M = \text{W}, \text{Mo}$), varying between the range of 0 to 1. For ideal Debye-type behavior, the parameter $s = 1$, $A\omega^s$ is the ac conductivity encompassing all dispersion phenomena.

As is evident from the conductivity plots, two distinct regions can be observed at low and high frequencies. In the low-frequency range, the conductivity spectra exhibit nearly constant values, indicating that the charge transport can be described by a hopping model, as defined by the first term of the Jonscher equation. Ions successfully hop from one site to an adjacent vacancy in this area, causing long-range transitional motion of the ions, which contributes to the direct conductivity.

In contrast, the second part of the equation can be used to determine the asymptotic form of the conductivity, often referred to as the dispersive regime, at high frequencies.

This dispersion could be connected to the anticipated behavior of space charges, given that their influence diminishes with increasing temperatures and frequencies.³³ The proposed spectra demonstrate that this is not always the case, since there

are curves for which the increase is not observed, indicating the presence of space charges even at high frequencies.

In addition, we observed that the AC conductivity rises in tandem with temperature. Given that electrical conduction in the ceramic system is a thermally stimulated process, this indicates that a greater number of free charge carriers are generated at high temperatures, which promotes enhanced conductivity and suggests the properties of the semiconductor.³⁴

In fact, eqn (6) can be mentioned by the Almond–West relation:³⁵

$$\sigma_{ac}(\omega) = \sigma_{dc} \left(1 + \left(\frac{\omega}{\omega_h} \right) \right) \quad (8)$$

where ω_h , the frequency corresponding to the onset appearance of the dispersion region, is known as the hopping frequency of the charge carrier. With an increase in temperature, the ω_h shifts to a higher frequency range, indicating that the hopping frequency is thermally stimulated. This is stated in the following way:

$$\omega_h = \left(\frac{\sigma_{dc}}{A} \right)^{1/s} \quad (9)$$

Fig. 13(a) and (b) reveals the plot of $\ln(\omega_h)$ as a function of $(1000/T)$ relative to the conduction of grains for $\alpha\text{-CuMoO}_4$ and CuWO_4 compounds, respectively. It represents the continuity of the curve slope, referring to the absence of a phase transition within these samples at the studied temperature range. This was confirmed by the DSC (Differential Scanning Calorimetric) diagram.

These evolving processes exhibit an Arrhenius-like behavior with activation energies determined from a straight-line adjustment of the data points, roughly equal to $E_{a,g} = 1.36$ eV for the Mo-based compound and $E_{a,g} = 0.56$ eV for the W-based compound. In fact, these results differ from those found recently on the DC conductivity of grains (σ_g), demonstrating that the mobility of charge carriers in the studied compounds is not ensured by a straightforward hopping mechanism.³⁶

A suggested approach to understand the charge transfer mechanism in CuMO_4 ($M = \text{W}, \text{Mo}$) can be constructed by

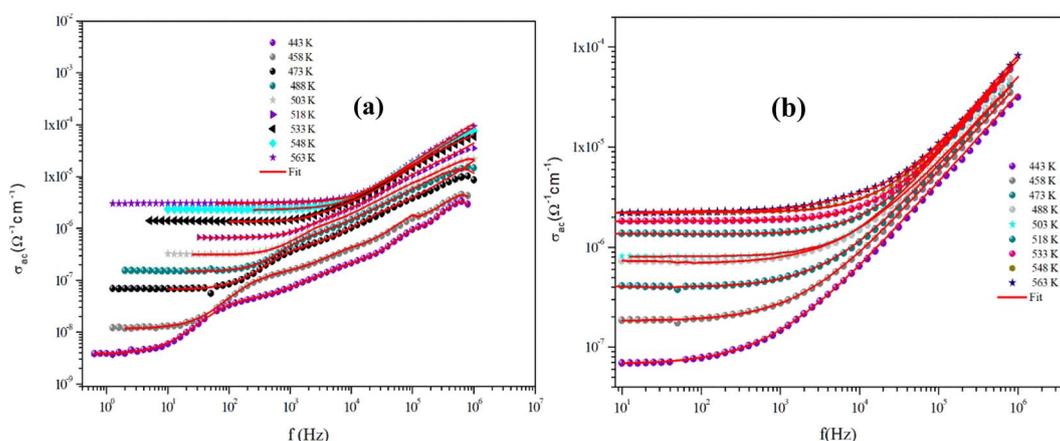


Fig. 12 Dependence of AC conductivity on the frequency at specific temperatures for $\alpha\text{-CuMoO}_4$ (a) and CuWO_4 (b).



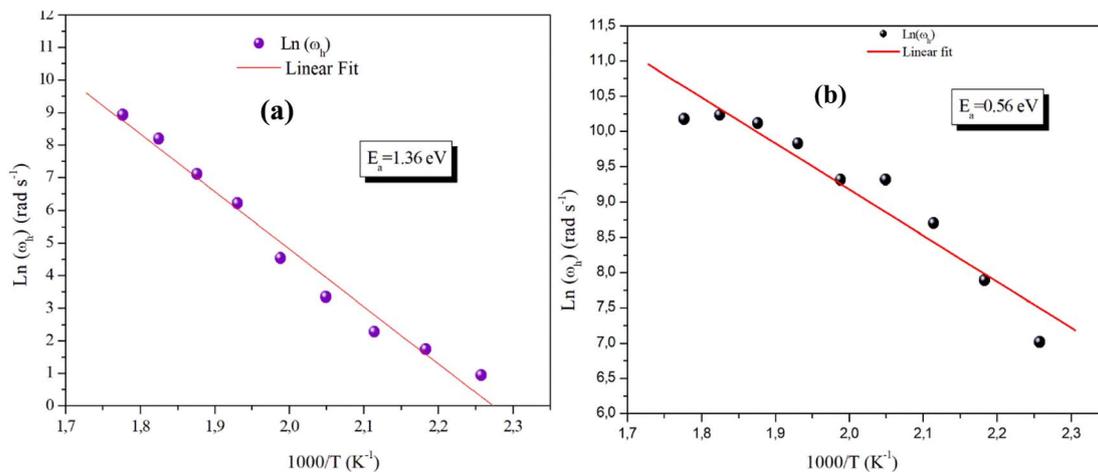


Fig. 13 Plot of hopping frequency ω_h against $1000/T$ for the compound $\alpha\text{-CuMoO}_4$ (a) and for CuWO_4 (b).

considering several theoretical models based on the behavior of s (T) with temperature.

According to the literature, various models have been proposed, such as the non-overlapping small polaron tunneling (NSPT) model put forth by Long,³⁷ the overlapping large polaron tunneling (OLPT) model proposed by Long,³⁸ the quantum mechanical tunneling (QMT) model put forth by Austin–Mott,³⁹ and the correlated barrier hopping (CBH) model, as introduced by Elliot.^{40,41} These diverse conduction models are based on two distinct processes: the quantum mechanical tunneling effect and the classical hopping of charge carriers across a barrier, or a combination of the two. Additionally, various hypotheses have been made about the atoms or electrons (or polarons) accountable for serving as carriers.

In this context, the variations of the exponent (s) with temperature in our systems are illustrated in Fig. 14(a) and (b).

In the temperature range of 443 K to 533 K, the parameter (s) for CuWO_4 shows an increase with temperature, indicating the presence of a correlated barrier hopping (CBH) mechanism. However, within the temperature range of 548 K to 563 K, the

exponent (s) displays a tendency to decrease with rising temperatures. Therefore, this result suggests that the non-overlapping small polaron tunneling (NSPT) model is a suitable mechanism to characterize electrical conduction. For $\alpha\text{-CuMoO}_4$, the asymptotic patterns of ' s ' decrease with increasing temperature, establishing a minimum at 473 K, then they start to increase with rising temperatures. Therefore, the OLPT model appears to be the most suitable model for characterizing the electrical conduction mechanism in the studied compound.

3.6. Complex modulus analysis

The electrical relaxation characteristics occurring in materials are frequently justified using the electrical modulus formalism.⁴² The electrode polarization and the relaxation frequency of conductivity are displayed by electrical modulus analyses. This formalization provides a more comprehensive understanding of charge transport processes, including conductivity relaxation, ion dynamics as a function of temperature and frequency, and the electrical transport mechanism. This formalism is complementary to the conductivity study.

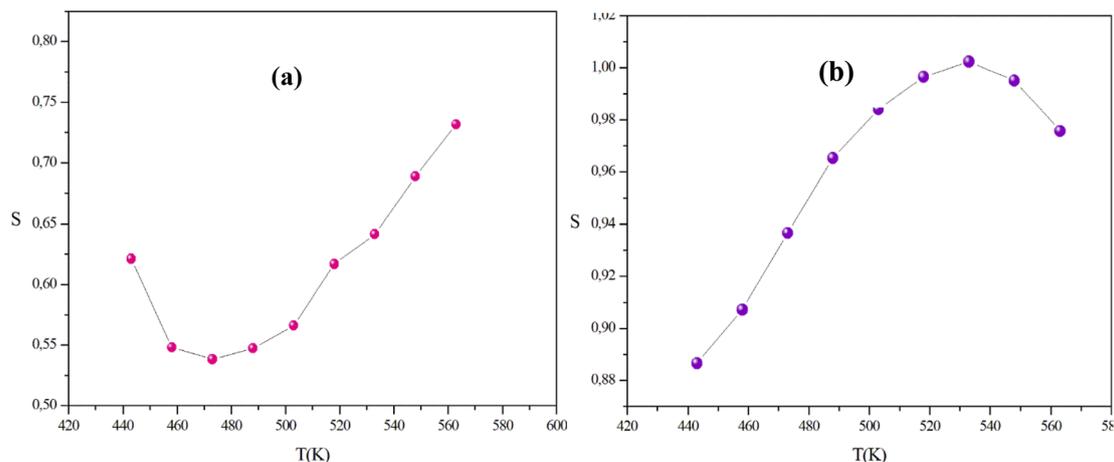


Fig. 14 The variation of the universal exponent s with the temperature of the $\alpha\text{-CuMoO}_4$ (a) and CuWO_4 (b) samples.



In fact, the dielectric modulus (M^*) is defined as:

$$M^* = M' + jM''$$

$$= M_\infty \left[1 - \int_0^\infty e^{-i\omega t} \left(-\frac{d\varphi(t)}{dt} dt \right) \right] \quad (10)$$

where $M' = \omega C_0 Z''$ and $M'' = \omega C_0 Z'$ are the real and imaginary parts of the modulus, respectively, and C_0 is the geometrical capacitance defined as $C_0 = \epsilon_0 S/e$ (where ϵ_0 denotes the permittivity of free space, S represents the area of the electrode, and e signifies the thickness). Additionally, $\varphi(t)$ represents the function that shows the time evolution of the electric field within the materials.

Fig. 15(a) and (b) shows the variation of the imaginary part of the modulus (M''), as a function of frequency for the two title compounds in the measured temperature range (443 K–563 K).

For the Mo-based compound, this graph displays two asymmetric relaxation peaks for each temperature. The influence of the grain boundary is demonstrated by the tiny peak observed at low frequencies. Whereas, the second peak found at high frequencies is connected to the relaxation process of grains. In

contrast, for the W-based compound, these plots show the appearance of a single peak in the modulus spectra, confirming the single relaxation phenomenon of the material studied. These results are in good agreement with those of the Nyquist plots.

Furthermore, it is evident that the illustrations of these figures reveal that the charge carriers are mobile over long distances in the frequency range below the maximum peak M''_{\max} . The area above is where charge carriers are spatially connected to potential wells and are only capable of localized motion within these wells while maintaining their mobility over short distances.

Additionally, the presence of these peaks in the modulus spectra explains the conductivity relaxation processes. With rising temperatures, the relaxation peak moves up to a higher frequency spectrum. The relaxation process is thus temperature-dependent and this indicates a correlation between the movement of the mobile ions as the temperature increases.

In fact, the charge carriers become more accelerated and thermally active. This results in a decrease in the relaxation time, an increase in the relaxation frequency, and a shift in the frequency of the relaxation peaks.⁴³ This behavior assumes that

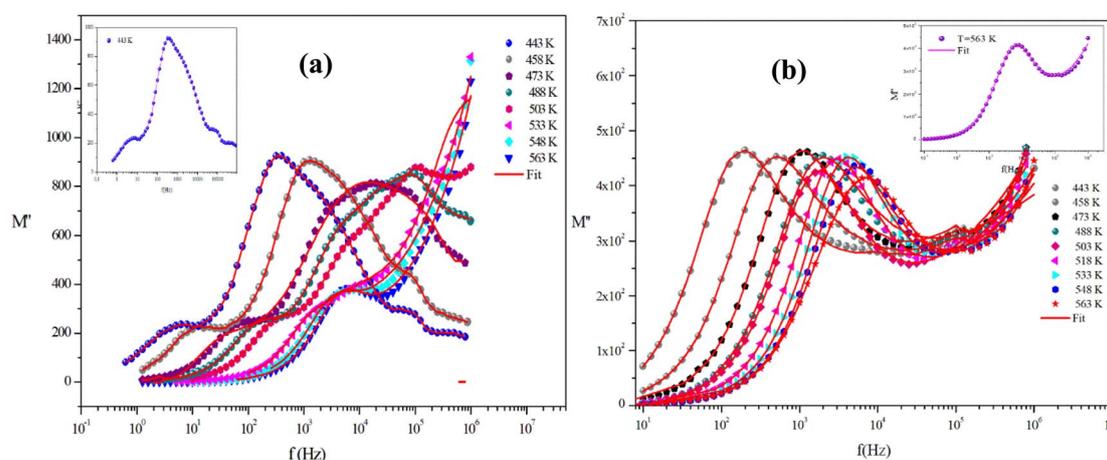


Fig. 15 The imaginary parts of the electric modulus as a function of frequency at several temperatures of α -CuMoO₄ (a) and CuWO₄ (b).

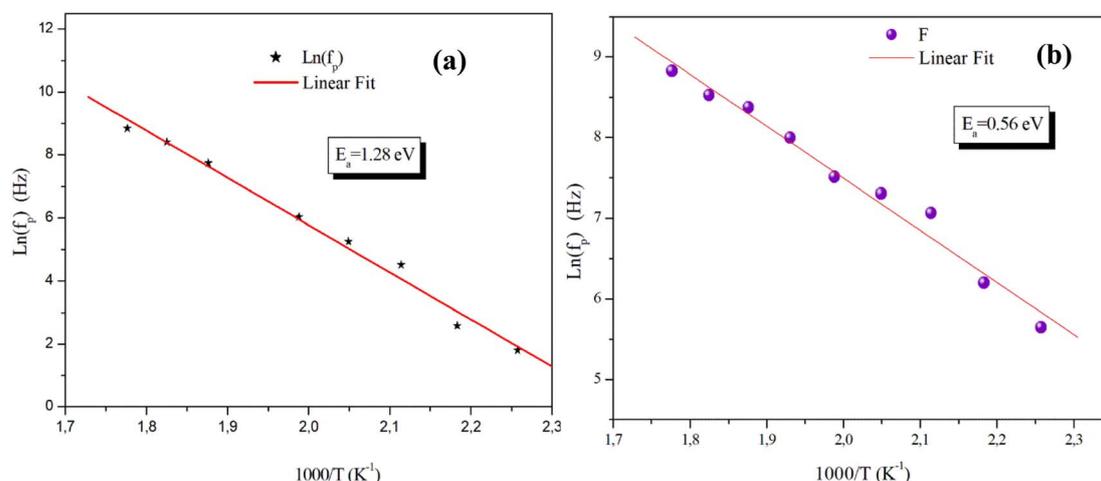


Fig. 16 Temperature dependence of the relaxation frequency f_p for the grains of α -CuMoO₄ (a) and CuWO₄ (b).



Table 4 Electrical and dielectric parameters of the grains for CuWO₄ and α -CuMoO₄ compounds

Sample	$\text{Ln}(\sigma_g T)$	$\text{Ln}(f_p)$
CuWO ₄	$E_a = 0.73$ eV	$E_a = 0.56$ eV
α -CuMoO ₄	$E_a = 0.60$ eV	$E_a = 1.28$ eV

the jumps of the charge carrier are taken into account and that the relaxation depends on the temperature.

Added to this, at lower frequencies, the values of the imaginary parts of M'' approach zero, showing that the electrode effect can be neglected in the modulus representation.

On the other hand, the temperature dependency of the characteristic relaxation frequency f_p is plotted in Fig. 16(a) and (b). It is clear that the relaxation frequency satisfies the Arrhenius law given by the following equation:

$$f_p = f_0 \exp\left(\frac{-E_a}{k_B T}\right) \quad (11)$$

where f_0 is the pre-exponential factor, E_a is the activation energy for the relaxation frequency, and k_B denotes the Boltzmann constant. The activation energy calculated by a linear fit to the data points is $E_a = 1.28$ eV for a Mo-based compound and $E_a = 0.56$ eV for a W-based compound.

The activation energies derived from $\text{Ln}(\sigma_g T) = f(1000/T)$ and $(\text{Ln} f_p) = f(1000/T)$ for the two title compounds are grouped in Table 4. It can be seen that the values of the activation energies obtained from the relaxation frequency and those obtained from the conductivity are different for the α -CuMoO₄ sample. This means that there is a partial blocking of the movements of the charge carriers by the grain boundaries.⁴⁴ However, the latter has a significant impact on the total conductivity values of the material.

Contrarily, in the case of the compound based on tungsten, the activation energy estimated from the relaxation frequency is very close to that obtained from the conductivity. This observation demonstrates that the total conductivity is equal to the grain conductivity rather than being influenced by the grain boundaries. This proves that conduction and relaxation are ensured by the same mechanism in the material.

4. Conclusions

Briefly, our study examined the impact of replacing tungsten ($r_{W^{6+}} = 0.60$ Å) with molybdenum ($r_{Mo^{6+}} = 0.59$ Å) on both the structural and electrical properties. Initially, we prepared powders of CuMoO₄ and CuWO₄ using a solid-state reaction method. The XRD diffractograms demonstrate that both CuMoO₄ and CuWO₄ crystallize in the triclinic system with the same space group $P\bar{1}$. From a structural standpoint, it is important to note that the replacement of tungsten with molybdenum results in a significant change in the lattice parameters, the volume, the distances between W and Mo, and the angles. Thermal analysis indicates the absence of any exothermic or endothermic peaks within the temperature range of 283 K to 483 K, which can be attributed to the absence of structural transitions in the two titled compounds.

An equivalent circuit for the electrochemical cell with both compounds has been proposed.

Furthermore, the study of dc conductivity implies that Mo⁺ conduction is preferable to W⁺ conduction as a result of its small mass and small ionic size. Additionally, it has been discovered that at various temperatures, the ac conductivity spectra response follows the Jonscher's power law. The variations of the exponent 's' as a function of temperature confirm that the charge transport mechanisms can be described by the non-overlapping small polaron tunneling (NSPT) model, the correlated barrier hopping (CBH) model for the sample of CuWO₄, and the OLPT approach for α -CuMoO₄. Eventually, the temperature variation of the M'' peak showed a thermally activated relaxation process.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors extend their appreciation to the Deanship of Scientific Research at King Khalid University for funding this work through a small group Research Project under grant number RGP.1/41/44.

References

- W. Xiusheng, F. Chao, C. Jufang, G. Chunye and L. Wei, Effect of W doping on phase transition behavior and dielectric relaxation of CuMoO₄ obtained by a modified sol-gel method, *Mater. Res. Express*, 2020, 7, 016309.
- K. Seevakan, Structural, morphological and magneto-optical properties of CuMoO₄, electrochemical nanocatalyst as supercapacitor electrode, *Ceram. Interfaces*, 2018, 44, 20075–20083.
- G. Steiner, R. Salzer and W. Reichelt, Temperature dependence of the optical properties of CuMoO₄ Fresenius, *J. Anal. Chem.*, 2001, 370, 731–734.
- N. Joseph, Glass-free CuMoO₄ ceramic with excellent dielectric and thermal properties for ultralow temperature cofired ceramic applications, *ACS Sustainable Chem. Eng.*, 2016, 4, 5632–5639.
- M. Naja, A. Abbasi, M. M. Farahani and V. H. Rodrigues, Synthesis, characterization and crystal structure of a copper molybdate coordination polymer as an epoxidation catalyst, *Inorg. Chim. Acta*, 2015, 433, 21–25.
- D. Dongwei, R. Lan, X. Wei, B. Richard, W. Huanting and T. Shanwen, Preparation of a hybrid Cu₂O/CuMoO₄ nanosheetelectrode for high-performance asymmetric supercapacitors, *J. Mater. Chem. A*, 2016, 4, 17749–17756.
- H. Ehrenberg, H. Weitzel, J. Garcia-Jaca, F. Rodriguez and D. Hernandez, Optical study of the piezochromic transition in CuMoO₄ by pressure spectroscopy, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, 61, 16497.
- M. Benchikhi, Characterization and photoluminescence properties of ultrafine copper molybdate(α -CuMoO₄)



- powders prepared via a combustion-like process, *Int. J. Miner., Metall. Mater.*, 2016, **23**, 1340–1345.
- 9 R. Kohlmuller and J. P. Faurie, Etude des systemes $\text{MoO}_3\text{-Ag}_2\text{MoO}_4$ et $\text{MoO}_3\text{-MO}$ (MCu, Zn, Cd), *Bull. Soc. Chim. Fr.*, 1968, **11**, 4379–4382.
- 10 H. Ehrenberg, H. Weitzel, H. Paulus, M. Wiesmann, G. Wltschek, M. Geselle and H. Fuess, Crystal structure and magnetic properties of CuMoO_4 at low temperature (β -phase), *J. Phys. Chem. Solids*, 1997, **58**, 153–160.
- 11 A. W. Sleight, High Pressure CuMoO_4 , *Mater. Res. Bull.*, 1973, **8**, 863–866.
- 12 R. Tali, V. V. Tabachenko, L. M. Kovba and L. N. Dem Öyanets, Kristallicheskaya struktura CuMoO_4 , *Zh. Neorg. Khim.*, 1991, **36**, 1642–1644.
- 13 J. Baek, A. S. Sefat, D. Mandrus and P. S. Halasyamani, A New Magnetically Ordered Polymorph of CuMoO_4 : Synthesis and Characterization of $\epsilon\text{-CuMoO}_4$, *Chem. Mater.*, 2008, **20**, 3785–3787.
- 14 H. Ehrenberg, M. Wiesmann, J. Garcia-Jaca, H. Weitzel and H. Fuess, Magnetic structures of the high-pressure modifications of CoMoO_4 and CuMoO_4 , *J. Magn. Magn. Mater.*, 1998, **182**, 152–160.
- 15 O. Yu. Khyzhun, V. L. Bekenev and Yu. M. Solonin, First-principles calculations and X-ray spectroscopy studies of the electronic structure of CuWO_4 , *J. Alloys Compd.*, 2009, **480**, 184–189.
- 16 J. Ruiz-Fuertes, D. Errandonea, A. Segura, F. J. Manjón, Zh. Zhu and C. Y. Tu, Growth, characterization, and high-pressure optical studies of CuWO_4 , *High Pressure Res.*, 2008, **28**, 565–570.
- 17 N. Wooster, The crystal structure of Molybdenum Trioxide, MoO_3 , *Zeitschrift für Kristallographie, Cryst. Mater.*, 2015, **18**, 504–5012.
- 18 J. B. Forsyth, C. Wilkinson and A. I. Zvyagin, The antiferromagnetic structure of copper tungstate, CuWO_4 , *J. Phys.: Condens. Matter*, 1991, **3**, 8433–8440.
- 19 M. Mumtaz, M. Naveed, B. Amin, M. Imran and M. N. Khan, Temperature dependent impedance spectroscopy of $(\text{Co}_3\text{O}_4)_x/\text{CuTi-1223}$ nanoparticles-superconductor composites, *Ceram. Int.*, 2018, **44**, 4351–4359.
- 20 K. Karoui, A. B. Rhaïem and F. Jemni, $\text{Li}_2\text{M}(\text{WO}_4)_2$ (M = Ni, Cu, Co): electrical, thermal, and optical properties, *Ionics*, 2021, **27**, 1511–1524.
- 21 D. Johnson, *ZPlot, ZView Electrochemical Impedance Software, Version 2.3b*, Scribner Associates, Inc., North Carolina, 2000.
- 22 I. Dakhlaoui, K. Karoui, F. Hajlaoui, M. Zaghrioui, N. Audebrand, M. Dallone and F. Jomni, A new supramolecular semiconductor palladium (II) complex $[(\text{CH}_3)_3\text{N}(\text{CH}_2)_3\text{Br}]_2\text{Pd}_2\text{Cl}_6$: structural study, optical and electrical properties, *New J. Chem.*, 2023, **47**, 8042–8049.
- 23 N. Chakchouk, K. Ben Brahim, M. Ben Gzaïel and A. Oueslati, Investigation of optical and electrical properties of the semiconducting $\alpha\text{-KZnPO}_4$ compound, *RSC Adv.*, 2022, **12**, 6831–6840.
- 24 S. Kour, H. Mahajan and R. Mukherjee, Influence of sintering temperature on impedance and modulus spectroscopy of nickel-substituted cobalt ferrite, *J. Mater. Sci.: Mater. Electron.*, 2023, **34**, 594–608.
- 25 M. M. Rahman, N. Hasan, M. A. Hoque, M. B. Hossen and M. Arifuzzaman, *Results Phys.*, 2022, **38**, 105610.
- 26 B. N. Parida, R. D. Piyush, R. Padhee and R. N. P. Choudhary, Synthesis and characterization of a tungsten bronze ferroelectric oxide, *Adv. Mater. Lett.*, 2012, **3**, 231–238.
- 27 L. Singh, L. Dhavala, R. Bhimireddi, A. A. Ansari, S. Kumar, V. Srivastava, R. N. Rai, Q. V. Le and Y. Lee, Low-cost flame synthesized $\text{La}_{2/3}\text{Cu}_3\text{Ti}_4\text{O}_{12}$ electro-ceramic and extensive investigation on electrical, impedance, modulus, and optical properties, *Cer. Inter.*, 2023, **49**, 21795–21803.
- 28 K. Karoui, A. Mahmoud, A. Ben Rhaïem and F. Boschini, Electrical and electrochemical properties of $\text{Li}_2\text{M}(\text{WO}_4)_2$ (M = Ni, Co and Cu) compounds, *RSC Adv.*, 2019, **9**, 6785–6792.
- 29 M. Ganaie, S. Ahmad, S. Islam and M. Zulfequar, Dc, Ac Conductivity and Dielectric Analysis of SeTe Alloy, *Adv. Sci. Lett.*, 2014, **20**, 1360–1363.
- 30 B. Louati, F. Hlel and K. Guidara, *J. Alloys Compd.*, 2009, **486**, 299–303.
- 31 A. K. Jonscher, *Dielectric Relaxation in Solids*, Chelsea Dielectric Press, London, 1983, p. 380.
- 32 R. Ltaoui, A. Triki, S. Hcini, A. Oueslati, S. Zemni and O. Kanoun, Conduction mechanisms and complex impedance analysis in $\text{La}_{0.6}\text{Sr}_{0.4}\text{FeO}_3$ ceramic, *J. Electron.*, 2023, **50**, 121–138.
- 33 A. Q. Abdullah, Optical and structural investigation of synthesized PVA/Pb Snanocomposites, *Chem. Mater. Res.*, 2015, **26**, 6939–6944.
- 34 R. N. Bhowmik and G. Vijayasri, *J. Appl. Phys.*, 2013, **114**, 223701–223708.
- 35 H. Nefzi, F. Sediri, H. Hamzaoui and N. Gharbi, *Mater. Res. Bull.*, 2013, **48**, 1978–1983.
- 36 N. Chakchouk, B. Louati and K. Guidara, Electrical properties and conduction mechanism study by OLPT model of NaZnPO_4 compound, *Mater. Res. Bull.*, 2018, **99**, 52–60.
- 37 M. Dult, R. S. Kundu, S. Murugavel, R. Punia and N. Kishore, *Phys. B*, 2014, **452**, 102–107.
- 38 A. R. Long, Frequency-dependent loss in amorphous semiconductors, *Adv. Phys.*, 1982, **31**, 553–637.
- 39 I. G. Austin and N. F. Mott, Polarons in crystalline and non-crystalline materials, *Adv. Phys.*, 1969, **18**, 41–102.
- 40 S. R. Elliot, *Adv. Phys.*, 1987, **36**, 135–217.
- 41 M. Krimi, K. Karoui, J. J. Suñol and A. B. Rhaïem, Phase transition, impedance spectroscopy and conduction mechanism of $\text{Li}_{0.5}\text{Na}_{1.5}\text{WO}_4$ material, *Phys. E*, 2018, **102**, 37–145.
- 42 C. T. Moynihan, L. P. Boesch and N. L. Laberge, Decay function for the electric field relaxation in vitreous ionic conductors, *J. Phys. Chem. Glas.*, 1973, **14122**, 122–125.
- 43 V. Thakur, A. Singh, R. Punia, S. Dahiya and L. Singh, *J. Alloys Compd.*, 2017, **696**, 529–537.
- 44 C. H. Hervoches, M. C. Steil and R. Muccillo, Synthesis by the polymeric precursor technique of Bi_2CoO_4 . $1V_0.9O_5$. 35 and electrical properties dependence on the crystallite size, *J. Solid State Sci.*, 2004, **6**, 173–177.

