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A mesoporous Mo and N Co-doped TiO_2 nanocomposite with enhanced photocatalytic efficiency

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This study reports the synthesis of a mesoporous Mo and N codoped anatase TiO_2 nanocomposite with many oxygen vacancies using a simple one-step hydrothermal method and subsequent calcination treatment. Both Mo and N were effectively co-incorporated into the anatase phase of TiO_2 without MoO_x phase segregation. The codoped catalyst demonstrated a mesoporous architecture with a surface area of $107.48 \text{ m}^2 \text{ g}^{-1}$ and a pore volume of $0.2974 \text{ cm}^3 \text{ g}^{-1}$. X-ray photoelectron spectroscopy confirmed that both Mo and N dissolved in the TiO_2 lattice and created induced oxygen vacancies. The interaction of the dopants (Mo and N) and oxygen vacancies clearly affected TiO_2 crystal formation. Photocatalytic performance of the nanocomposite was investigated in terms of the decomposition of methyl orange at a concentration of 50 mg L^{-1} in an aqueous solution. The results revealed a significant methyl orange degradation of up to 99.6% after 30 min irradiation under a UV light. The impressive performance of the nanocomposite is assigned to the synergetic effect of important factors, including the co-doping of metallic (Mo) and non-metallic (N) elements, oxygen vacancy defects, bandgap, crystallite size, mesoporous structure, and BET surface area.

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1. Introduction

Fabrication of nanostructured semiconductors is gaining considerable attention owing to their ability to absorb solar light radiation and their promising implications in environmental remediation *via* photocatalytic redox reactions.^{1–5} Currently, titanium dioxide (TiO_2) has been popularly utilized as a photocatalyst for the degradation of pollutants and energy conversion owing to its excellent advantages, including high refractive index, superior photocatalytic efficiency, low cost, high thermal and chemical stability, and non-toxicity.^{6–9} However, the wide bandgap (3.2 eV for anatase) of TiO_2 can only be activated using ultraviolet (UV) light, which has limited its practical applicability in visible-light-driven chemical reactions^{10,11} in addition to the fast recombination rate of photo-generated charge carriers.^{10,11}

Therefore, considerable efforts have been devoted toward widening the utilization scope of TiO_2 not only *via* extending the

photosensitivity of TiO_2 to the visible light region but also by diminishing the recombination rate of charge carriers.^{4,7,11} Doping or codoping TiO_2 with heteroatoms (metals or/and non-metals) is a promising strategy for narrowing its bandgap and thereby increasing the photocatalytic performance.^{4,7,11} Nitrogen is the most promising anionic dopant for TiO_2 because the size and ionization energy of both nitrogen and oxygen are almost identical.^{12,13} Moreover, nitrogen narrows the TiO_2 bandgap *via* coupling its 2p states with the 2p states those of TiO_2 oxygen, thus forming electronic levels above the valence band of O 2p.^{10,12,13} Note that doping TiO_2 with a metal having a higher oxidation state (especially Molybdenum; Mo^{6+}) than that of the parental Ti^{4+} atom would increase the separation and transfer of the photogenerated charge carriers.¹⁴ Molybdenum has a partially filled d electron shell and a tunable valence state, which make it a good doping candidate for fine-tuning the catalytic efficiency of a material.^{15,16} Furthermore, Mo is widely used as a TiO_2 dopant as it is non-toxic and inexpensive with good solubility in the anatase phase.¹⁴ Moreover, the similar radii of Mo^{6+} ions (0.062 nm) and Ti^{4+} ions (0.068 nm) facilitate effective Mo^{6+} substitution at Ti^{4+} sites with minimal or no lattice distortion.^{14,17–22} As for charge balancing, the Mo^{5+} species is usually produced (though Mo^{6+} is used as the doping precursor); otherwise, Ti^{3+} species or/and oxygen vacancies are formed.²³ Oxygen vacancies are the most popular type of TiO_2 intrinsic defects and are assumed to be the most active regions

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in TiO_2 .¹³ The presence of both oxygen-deficient regions and intrinsic Ti^{3+} surface defects induce the formation of new mid-bandgap levels that are located below the conduction band and act as donor or acceptor electronic levels.^{13,14}

Though mono-doping can diminish the TiO_2 bandgap to some extent, it suffers from the presence of recombination centers and the creation of tightly localized impurity bands inside the bandgap, which reduce carrier mobility.^{17,22} Thus, co-doping with a metal-cation and a nonmetal-anion has been reported to minimize the recombination centers *via* passivating the defect bands and preventing them from serving as recombination centers.^{4,17,24} Many studies demonstrated the remarkable enhancement of TiO_2 photoactivity *via* Mo and N co-doping and the shift in its light absorption ability to the visible light range.^{17,18,22,24,25} However, the development of a facile approach to fabricate nanostructured semiconductor photocatalysts with visible-light photocatalytic efficiency remains a challenge.

This study aims to promote the photoresponse of anatase titania to visible light and thereby its photocatalytic performance by a facile hydrothermal process that achieves Mo and N codoping and oxygen deficiency formation. The prepared nanocomposite was characterized for elucidating the noticeable improvement in photoactivity. The photocatalytic activity of the obtained nanocomposite was evaluated in the photo-degradation of methyl orange solution under UV light irradiation. Relevant reasons for its superior photocatalytic performance are outlined.

2. Experimental

2.1. Chemicals

The starting materials, including commercially available ammonium molybdate tetrahydrate (AMT: $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24} \cdot 4\text{H}_2\text{O}$; 99.9%), ammonium oxalate ($\text{C}_2\text{H}_8\text{N}_2\text{O}_4$; 99.5%), ascorbic acid ($\text{HC}_6\text{H}_5\text{O}_6$; 99%), titanium(III) chloride (TiCl_3 ; 99%), sodium lauryl sulfate (SLS: $\text{NaC}_{12}\text{H}_{25}\text{SO}_4$; 99%), ethanol ($\text{C}_2\text{H}_5\text{OH}$; 99.9%) and methyl orange ($\text{C}_{14}\text{H}_{14}\text{N}_3\text{NaO}_3\text{S}$; 85%), were obtained from Sigma-Aldrich (Missouri, United States). Double distilled water (DIW) was used for sample preparation.

2.2. Photocatalyst preparation

Porous (Mo + N) codoped TiO_2 nanocomposite powders with carbon-modified surfaces were synthesized using a hydro-thermal process. A mixture of 0.1 g AMT, 0.5 g ammonium oxalate, and 0.5 g ascorbic acid was homogeneously blended in a manual mill. The mixture was dropped into a solution of 15 mL titanium(III) chloride, 0.2 g SLS, and 75 mL DIW and vigorously stirred for 45 min until a gel was formed. The fabricated gel was transferred to a Teflon-lined stainless steel autoclave (100 mL), sealed, and maintained in an oven at 155 °C for 40 h. The autoclave was then naturally cooled to room temperature in the air. The obtained precipitates were filtered, washed with DIW and ethanol several times and dried at 65 °C for 5 h under a vacuum. Finally, the fabricated samples were calcined at 500 °C for 3 h in the air by heating at a rate of 2 °C min^{-1} . The final product was defined as a Mo-doped TiO_2

photocatalyst (MT1). Similar procedures were repeated by varying the content of ammonium molybdate tetrahydrate to 0.2 and 0.3 g for fabricating catalysts with different Mo doping amounts, and these are defined as MT2 and MT3, respectively. Furthermore, undoped TiO_2 was prepared by the same method without adding the Mo precursor and used as a reference.

2.3. Characterizations of the synthesized photocatalyst

The phase structure of synthesized powders was identified using an X-ray diffractometer (XRD, Panalytical X'Pert Pro) with $\text{CuK}\alpha$ radiation (wavelength of 0.15406 nm) operated at 45 kV, 40 mA, and 25 °C. The spectra were recorded in the range of 10°–80° with a step size of 0.013. The surface microstructure and elemental composition were observed using a scanning electron microscope (SEM, JEOL JSM, Japan) equipped with an energy-dispersive X-ray (EDX) detector. The N_2 adsorption-desorption isotherm measurement was conducted at 77 K using the ChemBET Pulsar TPR/TPD instrument. X-ray photoelectron spectroscopy (XPS) was performed using equipment from ThermoFisher Scientific, USA system (USA) to investigate the surface chemical composition. The elemental analysis was performed using inductively coupled plasma-optical emission spectrometry (ICP-OES, LabTech, USA). EIS measurements of both un-doped and doped titanium oxide were conducted by applying an AC signal with a 10 mV amplitude in the frequency range of 0.1 Hz–50 M Hz at the open-circuit potential.

2.4. Photocatalytic activity

The photocatalytic efficiency of the as-prepared samples was evaluated using a Hitachi U-3900H double-beam UV-visible spectrophotometer in the region of 250–750 nm. The photocatalytic activity of the samples was estimated in terms of methyl orange dye (MO, a model pollutant) degradation under UV-Vis light illumination (wavelength range: 200 to 900 nm) using a 350 nm UV lamp. The MO photodegradation experiments were performed in the air at room temperature inside a sealed and black-painted wooden box with a UV lamp installed on the top. The degradation test was performed using the batch method under continuous stirring using a magnetic stirrer. In a typical run, 10 mg of the synthesized photocatalyst powder was added to a beaker containing 10 mL of MO solution at 50 mg L^{-1} . The mixture was sonicated under UV light for an appropriate time, and the reaction mixtures were collected at regular intervals by instant filtration. The decolorization of the dye solution was determined by measuring its absorbance at 460 nm on a UV-visible spectrophotometer as a function of exposure time.

3. Results and discussion

3.1. Crystal phase analysis of the synthesized photocatalysts

Herein, we present the analysis of only one of the doped TiO_2 samples, namely MT3, as it exhibited the best photocatalytic activity compared with other samples. The ICP analysis confirmed that the Mo concentration per gram of TiO_2 was 83 mg (8.3 wt%). The phase spectrum of the synthesized Mo-N



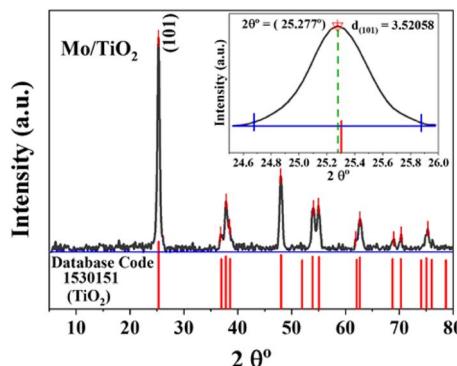


Fig. 1 The XRD pattern of the as-prepared (Mo + N) codoped TiO₂ NPs.

codoped TiO₂ powder is shown in Fig. 1. All the identified peaks perfectly refer to the anatase phase only (database code no: 1530151), which reveals a highly crystalline structure.^{21,26} The high crystallinity is beneficial to increasing the photocatalytic efficiency of the powder. The characteristic peaks at the approximate 2θ values of 25.227° (101), 37.827° (004), 38.477° (200), 54.027° (105), 54.977° (211), 62.677° (204), 68.977° (116), 70.327° (220) and 75.277° (215) correspond to the anatase phase (database code: 1530151).^{21,26} Remarkably, the diffraction pattern of the doped TiO₂ sample was fairly identical to pure TiO₂ (database code: 1530151) but with a slight shift in peak positions, as shown in the inset of Fig. 1. Moreover, no significant peaks of the free Mo phases were detected. This indicates that the Mo⁶⁺ ions were incorporated into the TiO₂ crystal lattice by either substituting the Ti⁴⁺ ions or occupying vacant sites in the lattice.^{22,27,28} Since the radii of the Mo⁶⁺ ions (0.062 nm) and Ti⁴⁺ ions (0.068 nm) are very similar, Mo can easily substitute Ti⁴⁺ and occupy its lattice position rather than an interstitial position. This leads to changes in the crystal plane distance and lattice parameters (lattice contraction), as previously reported.^{14,17-20} Note that Mo incorporation might not be detected by XRD as it could be below the detection limit, as reported previously.²⁰ The changing color of TiO₂ with the Mo content confirmed Mo incorporation. Lattice distortion owing to the dopants (ε) was calculated based on the following formula,^{29,30} $\varepsilon = \beta/(4 \tan \theta)$, where β is the full width at half maximum (FWHM) of the main peak and θ is the reflection angle. The Mo-N codoped TiO₂ composite revealed a lattice distortion of 0.3595, which may lead to an increase in lattice defects and assist carrier recombination.^{28,31} Moreover, (101) was the dominant plane, which signifies that the Mo-N codoped TiO₂ photocatalyst would preferably exhibit a (101) aligned texture. The d-spacings or interplanar spacings between the lattice fringes (d_{hkl}) were calculated from X-ray diffraction data using the Bragg Equation $d = (n \lambda)/(2 \sin \theta)$,³² where d indicates the distance between the atomic layers, λ is the wavelength of the incident X-ray beam, and n is an integer = 1. All d_{hkl} of Mo-N codoped TiO₂ matched well with those of TiO₂ in the database code, except for d_{101} , which was 3.52058 nm. This fluctuation in lattice characteristics might reflect a change in oxygen

vacancies, which are often related to the incorporation of impurities.³³ The average crystallite size (D) was evaluated by the Scherrer formula $D = (K \lambda)/(\beta \cos \theta)$,³⁴ where K is the shape factor (here 0.89), λ is the X-ray radiation wavelength (0.15406 nm for Cu K α), θ is the reflection angle and β is the FWHM. Mo-N codoped TiO₂ exhibited a crystallite size of 18.144 nm. Using the full width at half maximum (FWHM) of the (101) diffraction peak, the (101) interplanar distance in Mo-N codoped TiO₂ was found to be 0.352 nm, which is somewhat greater than that of pure TiO₂ anatase (0.35165 nm), indicating that the N and Mo atoms were implanted in the TiO₂ lattice.¹⁷

3.2. Morphological analysis of the as-synthesized photocatalysts

Fig. 2 shows the surface morphology and EDX elemental mappings of the as-prepared photocatalysts. The SEM images in Fig. 2(a and b) reveal that the as-prepared nanocomposite had a pseudo-spherical morphology with numerous caves and mounds. This can be attributed to the slight agglomeration of the nanocrystals. The elemental mappings of the Mo and N codoped TiO₂ NPs in Fig. 2 (c-f) indicate the uniform distribution of Ti, O, and Mo in the as-prepared composites. The EDX elemental spectrum in Fig. 2(g) indicates that the as-prepared photocatalyst NPs comprised Ti, O, and Mo without any impurities, and the Mo-doping content was 3.93 atomic%. The N peak was not particularly noticeable probably because of the low N content in TiO₂.^{22,35}

3.3. Textural properties of the as-synthesized photocatalysts

The surface area, type of adsorption isotherms, and the pore sizes/shapes are important factors that determine the catalytic performance. Therefore, the textural features of the as-prepared Mo and N codoped TiO₂ NPs were investigated using the N₂ sorption technique at a low temperature of 77 K. The nitrogen desorption results, as shown in Fig. 3(a), demonstrated a typical type IV isotherm with a type H3 hysteresis loop, indicating the formation of a mesoporous structure with slit-shaped pores as a result of the stacking of plate-like particles.^{36,37} Furthermore, the corresponding pore size distribution was calculated using the Barrett-Joyner-Halenda (BJH) method from the desorption branch of the isotherm, as shown in Fig. 3(b). The BET specific surface area (S_{BET}) was 107.48 m² g⁻¹, and the total pore volume was 0.2974 cm³ g⁻¹. The measured values indicate that the as-prepared material has excellent surface and mesoporous properties and thereby will lead to high photocatalytic activity. Moreover, the incorporation of a dopant is coupled with the emergence of various defects in the TiO₂ crystal structure. The accumulation of defects, such as oxygen vacancies, assists the generation of pores, thus boosting the surface area, providing more active sites, and consequently enhancing the photocatalytic activity of the prepared photocatalyst.^{11,38,39}

3.4. XPS analysis and electronic structure

The XPS analysis was used to explore the bonding interaction and the oxidation states of the constituent elements in undoped TiO₂ and the (Mo + N) codoped TiO₂ nanocomposite. Whenever



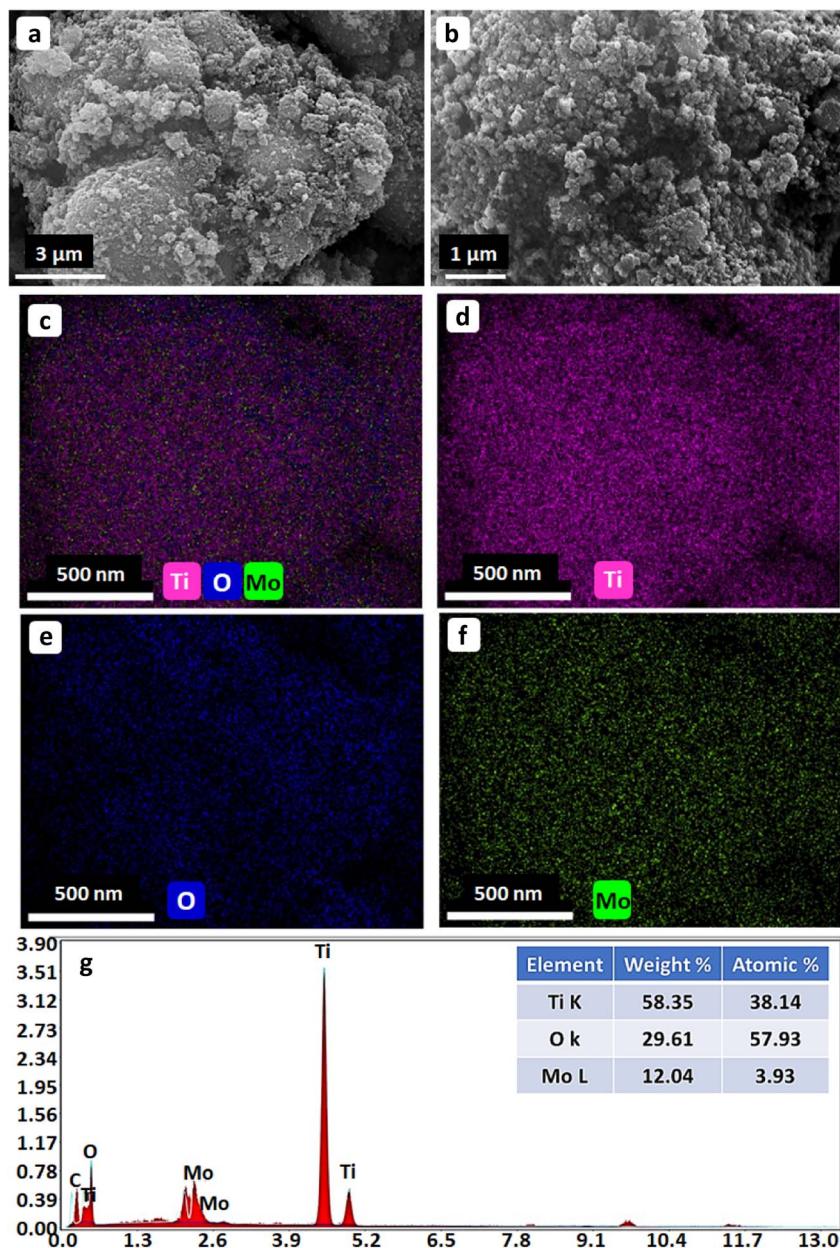


Fig. 2 The SEM images of the as-prepared (Mo + N) codoped TiO_2 NPs at (a) low magnification and (b) high magnification; (c–f) the EDX elemental mappings and (g) EDX spectrum of the as-prepared (Mo + N) codoped TiO_2 NPs.

isomorphic substitution takes place owing to the codoping of Mo and N, the extra positive charges offered by the Mo^{6+} species would be compensated using different means. This includes the creation of oxygen vacancies, the generation of (reduced) Mo^{6+} ions, and the reduction of Ti^{4+} ions to Ti^{3+} ions (intervalence charge transfer (IVCT)).^{21,23,40} The XPS spectra of as-prepared nanocomposites clearly confirmed the existence of Ti, Mo, O, N, and C, as shown in Fig. 4(a), indicating the effective formation of the Mo and N codoped TiO_2 nanocomposite. The presence of C could be attributed to the contamination of adventitious carbon. The Mo content (atomic ratio) was 4.96%, which is higher than the theoretical value of 3.6%, revealing the surface enrichment of the Mo ions.^{21,23,31} Furthermore, the

intensity and width of the N 1s peak of the doped sample were higher than those of bare TiO_2 . This is presumably related to the crystal lattice distortion caused by Mo doping, which facilitates the doping and solubility of N atoms in the structure.^{22,25}

Compared with undoped TiO_2 , in Fig. 4(b), (Mo + N) codoping slightly reduces the binding energies of the $\text{Ti}2\text{p}_{1/2}$ and $\text{Ti}2\text{p}_{3/2}$ orbitals. This reveals a relative drop in Ti valence, that is, the conversion of Ti^{4+} ions to Ti^{3+} accompanied by oxygen loss (oxygen vacancies) from the TiO_2 surface and substitution with N and Mo.⁴⁰ Fig. 4(c) shows the high-resolution XPS scan over the $\text{Ti}2\text{p}$ peak of the (Mo + N) codoped TiO_2 nanocomposite. It reveals two distinct peaks at about 464.17 and 458.43 eV, which are indexed to the $\text{Ti}2\text{p}_{1/2}$ and



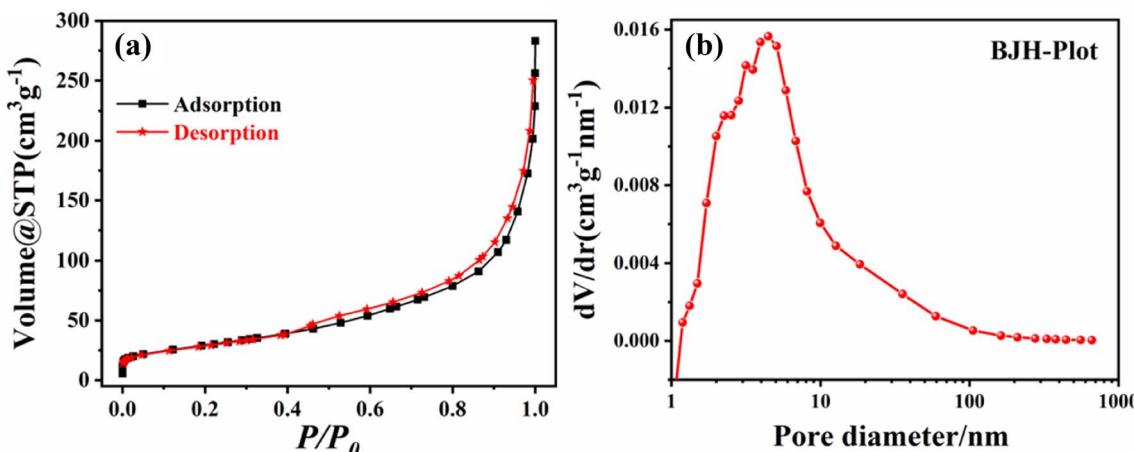


Fig. 3 (a) The N_2 adsorption–desorption isotherm and (b) BJH pore-size distribution (BJH plot) of the as-prepared (Mo + N) codoped TiO_2 NPs.

$\text{Ti}2\text{p}_{3/2}$ orbitals of Ti^{4+} (TiO_2), respectively.^{41–43} Moreover, the split of the 5.7 eV peak between the $\text{Ti}2\text{p}$ doublets clearly confirms the presence of Ti^{4+} ions in the TiO_2 lattice.^{43–46} Furthermore, some peaks revealing the presence of the Ti^{3+} and Ti^{2+} oxidation states were observed.^{47,48} The peaks at 463.16 and 457.93 eV are attributed to the $\text{Ti}2\text{p}_{1/2}$ and $\text{Ti}2\text{p}_{3/2}$ orbitals of Ti^{3+} present in the TiO_2 lattice,^{47,48} which confirms the conversion of Ti^{4+} to Ti^{3+} upon Mo-doping *via* intervalence charge transfer (IVCT). In this process, a charge compensation mechanism occurs; the reduction of Ti^{4+} to the less stable Ti^{3+} ion is associated with Mo⁵⁺ oxidation to the more stable Mo⁶⁺ phase.^{21,40} The difference in the redox potentials of $\text{Ti}^{4+}/\text{Ti}^{3+}$ (0.1 eV) and Mo⁶⁺/Mo⁵⁺ (0.4 eV) assists the capture of the electrons photo-generated by the Mo⁶⁺ ions. Consequently, the recombination rate of the e^-/h^+ pairs is reduced, leading to enhancement in the photocatalytic activity.^{25,49}

Mo $3\text{d}_{5/2}$ and Mo $3\text{d}_{3/2}$ peaks with higher binding energy were observed at 232.43 and 235.53 eV, as shown in Fig. 4(d). These are attributed to the Mo⁶⁺ oxo-species with a 3.1 eV spin-orbital doublet splitting.^{27,50} Furthermore, the signals at lower binding energies of 231.69 and 235 eV are ascribed to the Mo $3\text{d}_{5/2}$ and Mo $3\text{d}_{3/2}$ of Mo⁵⁺ oxo-species, respectively.^{19,41,51,52} Therefore, the Mo⁶⁺ and Mo⁵⁺ states of Mo coexist at an atomic proportion of 71.4% and 28.6%, respectively. The presence of Mo⁵⁺ ions is a sign of oxygen deficiency.¹⁹ The O 1s XPS spectrum could be deconvoluted into only two signals, as shown in Fig. 4(e). The higher signal centered at 529.68 eV corresponds with the oxygen within the TiO_2 crystal lattice (Ti^{4+}O).^{41,53} The lower energy signal at 530.05 eV is presumably attributed to the adsorbed oxygen in the oxygen-deficiency regions (owing to lattice distortion and the porous structure).^{19,27,31,41} Herein, the as-prepared (Mo + N) codoped TiO_2 NPs exist in an oxygen-deficient state owing to the inclusion of the dopant cations. Thus, one Ti atom requires two O atoms, whereas one Mo atom requires three O atoms, as well as due to the presence of the Mo⁵⁺ ions.^{19,53} Fig. 4(e) and its inset show the comparison between doped and undoped TiO_2 . It reveals higher O adsorption on the surface of the (Mo + N) codoped TiO_2

nanocomposite compared with the bare TiO_2 surface. The lack of oxygen on the surface can be complemented by adsorbing more oxygen, which is considered an advantage for photocatalytic degradation.^{19,41} Furthermore, the N 1s peak was located at 399.5 eV (Fig. 4(f)), suggesting that the nitrogen atoms were interstitially incorporated in the TiO_2 lattice and formed Ti–N–O or Ti–O–N oxynitride linkages.^{24,54–57} This observation reveals that metal doping enhances N solubility in the TiO_2 lattice.²² Moreover, the existence of the N dopant is accompanied by the creation of oxygen vacancies.⁵⁸

The estimated oxidation-reduction potentials of $\text{Ti}^{4+}/\text{Ti}^{3+}$ and Mo⁶⁺/Mo⁵⁺ are 0.1 and 0.4 eV²⁵, respectively. This indicates the greater ability of Mo⁶⁺ to catch photogenerated electrons, get distinct carriers, and boost the photocatalytic efficiency. Therefore, the photogenerated electrons can be captured, the rate of electron–hole recombination can be minimized and the photocatalytic reaction can be accelerated by injecting a suitable amount of the dopants.²⁵

3.5. Electrochemical impedance spectroscopy (EIS)

EIS is a powerful technique used to probe the various kinetic processes occurring at the electrode–electrolyte interface. Electrochemical impedance spectroscopy across a wide frequency range was conducted to extensively understand the electrode kinetics. The Nyquist plots (Fig. 5) of both undoped TiO_2 and Mo-doped TiO_2 samples displayed characteristic signatures of charge transfer and ion diffusion processes. Note that the diameter of the semicircular arc was smaller for Mo– TiO_2 , indicating faster charge transfer kinetics compared to pure TiO_2 . This enhancement arises from the oxygen vacancies introduced in the material when the Ti^{4+} ions are partially substituted by Mo⁶⁺ during doping.^{59,60} The oxygen vacancies compensate for the charge imbalance and dramatically increase the electrical conductivity. By providing additional pathways for charge transport, the oxygen vacancies facilitate faster charge transfer across the Mo– TiO_2 sample, thereby reducing the cumulative internal resistance.



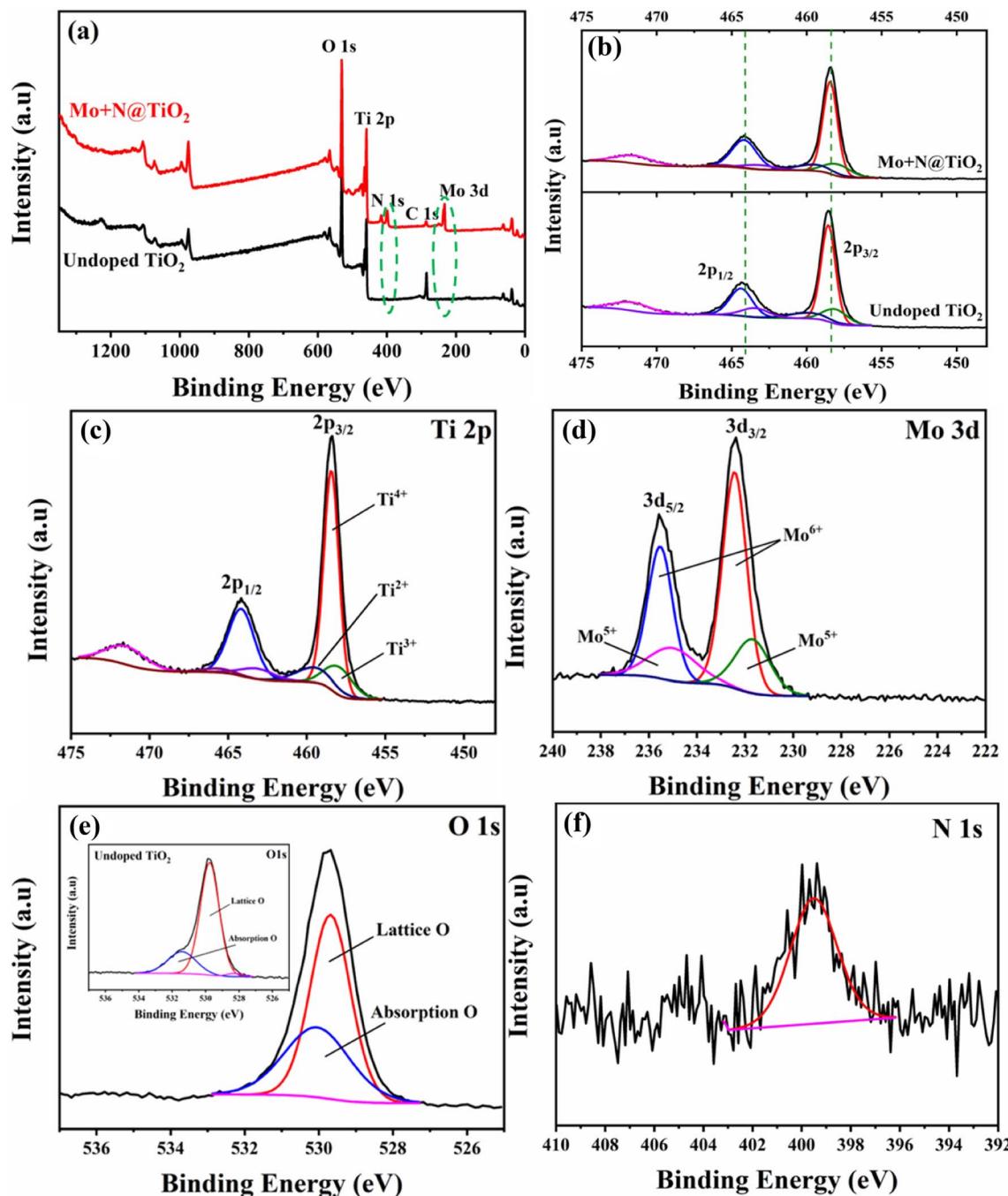


Fig. 4 (a) The XPS survey spectra and (b) Ti2p XPS spectra of undoped TiO_2 and the (Mo + N) codoped TiO_2 nanocomposite; the XPS spectra of the (c) Ti2p, (d) Mo3d, (e) O 1s, (f) N 1s core levels of the (Mo + N) codoped TiO_2 nanocomposite. The inset in Fig. 4(e) shows the O 1s spectrum of undoped TiO_2 .

3.6. Photocatalytic activity of the synthesized photocatalysts

The photocatalytic activity of the as-prepared (Mo + N) codoped TiO_2 nanocomposite was evaluated based on the degradation of MO in an aqueous solution under UV-light irradiation. The change in MO concentration at λ_{max} (482 nm) was monitored using a UV-vis spectrophotometer. Aqueous suspensions of undoped and doped TiO_2 nanopowders were used for the UV-vis absorption studies. Fig. 6 shows the changes in the absorption

spectra of MO in the presence of undoped TiO_2 and the (Mo + N) codoped TiO_2 nanocomposite. In the dark condition, there was no significant change in MO concentration; consequently, no significant change was detected in the adsorption of MO on the surface of the as-prepared catalysts. This indicates that the decolorization was primarily caused by the photocatalytic degradation of MO rather than its physiochemical adsorption on the surface of the nanostructured hybrid. Moreover, in the absence of a catalyst, <0.3% of MO was degraded after 30 min

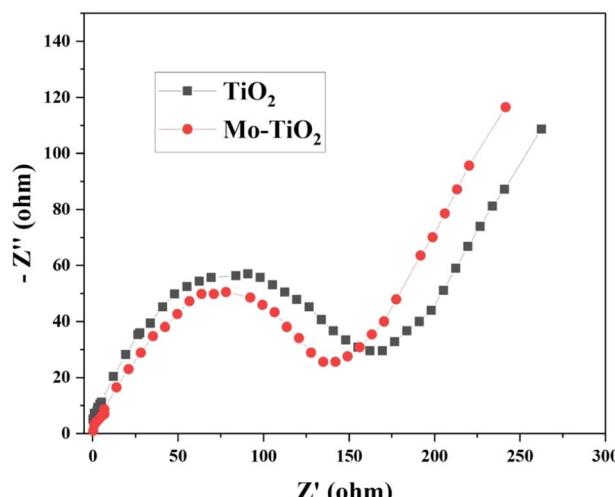


Fig. 5 The electrochemical impedance spectra (EIS) of the undoped TiO_2 and $\text{Mo}-\text{TiO}_2$ samples.

under UV irradiation, which demonstrates the high photo-stability of MO under UV light irradiation and that the auto-degradation of MO can be ignored, as shown in Fig. 6. After a few minutes of UV-irradiation, the addition of 10 mg of the fabricated photocatalysts to 10 mL of 50 mg L^{-1} MO leads to a significant decrease in absorption intensity owing to the decolorization of the dye. This reveals the improvement in the photocatalytic activity of the (Mo + N) codoped TiO_2 catalysts with increasing Mo content in the following sequence MT1 \prec MT2 \prec MT3. Furthermore, the photocatalytic activity of the TiO_2 catalyst was highly enhanced by Mo/N co-doping compared with undoped TiO_2 . The improvement in TiO_2 photocatalytic performance is related to the changing physical properties of the nanoparticles with Mo and N doping, producing alterations in the crystal structure and energy bandgap, as well as elemental composition. The ionic radius of Mo^{6+} is 0.062 nm, which is almost identical to that of Ti^{4+} (0.068 nm); therefore the Mo^{6+} ions can substitute Ti^{4+} in the TiO_2 anatase crystal lattice, which would minimize lattice distortion.^{22,61,62} Mo

doping would generate energy states within the bandgap of TiO_2 , which presumably shifts the optical absorption edge from the UV range to the visible light range. This consequently enhances light absorption and minimizes electron–hole recombination.^{61,63} Moreover, N can easily substitute O in the TiO_2 lattice owing to its comparable atomic size with O. The small ionization energy and high stability of N lead to the construction of an energy level above the VB of TiO_2 , which decreases the optical energy bandgap of TiO_2 and allows it to absorb light of higher wavelengths.^{64,65} Furthermore, it acts as a trapping center for the photogenerated electrons and suppresses the recombination rate of the photogenerated electrons and holes.^{64,65} This synergistic effect caused by the co-doping of metal and non-metal ions considerably enhances the photocatalytic activity of TiO_2 .

Note that no additional peaks were observed in the UV-vis spectra, which indicates that the photodegradation of the tested dye occurs without the formation of hazardous organic intermediates.

The pH of the solution considerably influences the catalyst surface charge and plays an important role in the photocatalytic

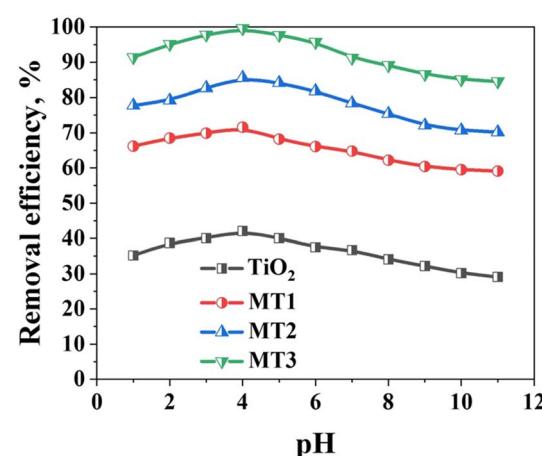


Fig. 7 The effect of pH on the degradation of MO in the presence of TiO_2 , MT1, MT2, and MT3 at the optimal condition.

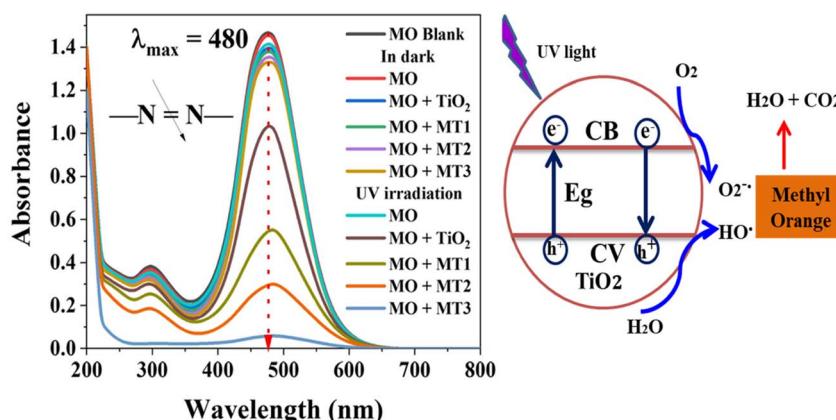


Fig. 6 The absorption spectra of the MO dye during the degradation process catalyzed by the as-prepared nanocomposites at $\lambda_{\text{max}} = 480 \text{ nm}$.



reaction, as well as the adsorption of the dye on the photocatalyst surface.^{66,67} Fig. 7 shows the effect of solution pH (1 to 11) on MO degradation. A fixed weight (10 mg) of the as-synthesized photo-catalysts and a constant 10 mL volume of the 50 mg L⁻¹ MO solution at room temperature were used in the experiment and irradiation for 30 minutes under a UV light. At pH = 4, MO was almost completely degraded (99.6%) within 30 min of the photoelectrocatalytic process. Moreover, the MO degradation efficiency apparently decreased with an increase in the pH value. The high MO removal efficiency under acidic conditions is related to the influence of the H⁺ ions since they can be adsorbed on the surface of the photocatalyst, making it positively charged. The positively charged surface allows the UV-generated e⁻ to migrate and interact with the adsorbed O₂ molecules, forming 'O₂. Furthermore, the positively charged surface might hinder the recombination of e⁻ and h⁺, additionally leading to OH⁻ formation through the interaction of h⁺ and H₂O.^{66,67} Furthermore, at lower pH, the MO molecules exist in the quinonoid form, and the degradation rate may further increase in the absence of the stable azo bond.⁶⁶ Moreover, the surface of TiO₂ is positively charged under acidic conditions, triggering TiO₂ to adsorb more negatively charged organic dye

molecules on the surface and thus enhancing the photocatalytic efficiency.^{66,67} However, in the alkaline condition, relatively lesser interactions would occur between the negatively charged TiO₂ and the organic dye anions, which decreases the amount of dye adsorbed on the surface and inhibits the photocatalytic process.^{66,67}

Fig. 8 shows the degree of MO degradation by the as-prepared (Mo + N) TiO₂ catalysts with different Mo contents as a function of reaction time. Pure TiO₂ demonstrates a degradation rate of 42.1% after 30 min, and the molybdenum-doped materials exhibited improved photocatalytic performance than undoped TiO₂. The quantity of MO degraded by MT1, MT2, and MT3 within the first 5 min was 19.8, 31.6%, and 51.6%, respectively. Moreover, MT3 revealed the highest efficiency of 99.6% when the contact time was increased to 30 min, whereas the MT1 and MT2 photocatalysts showed degradation rates of 71.7% and 85.6%, respectively. Herein, higher concentrations of the dopant molybdenum led to higher degradation.

The reaction kinetics was investigated to understand the photocatalytic MO dye degradation behavior of the (Mo + N) codoped TiO₂ nanocomposite. To properly compare the breakdown rates, the MO content was normalized by its value at the onset of light irradiation (C_t/C_0) and plotted *versus* t , as shown in Fig. 9. The time zero on the horizontal axis of the plot denotes the commencement of light irradiation, at which the C_t/C_0 value was set to 1 for all circumstances. The change in the MO adsorption peak value with time from 0 to 30 min in the presence of pure TiO₂, MT1, MT2, and MT3 was investigated with

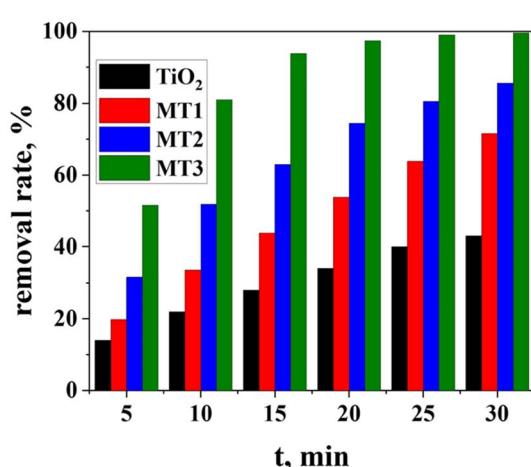


Fig. 8 The MO removal rates of the synthesized photocatalysts.

Table 1 The pseudo-first-order kinetic model of photocatalytic MO degradation by TiO₂, MT3, MT2, and MT1 at pH 4

Photocatalyst	Pseudo-first-order kinetic parameter	
	K	R ²
TiO ₂	1.83×10^{-2}	0.98116
MT1	4.09×10^{-2}	0.99536
MT2	6.405×10^{-2}	0.99728
MT3	1.87×10^{-1}	0.99892

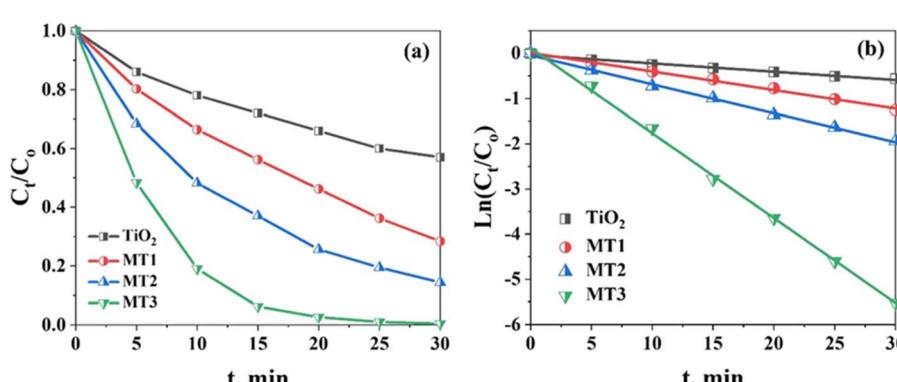


Fig. 9 (a) The MO degradation efficiency of TiO₂, MT1, MT2 and MT3; (b) the pseudo-first-order kinetics of MO photodegradation using TiO₂, MT1, MT2, and MT3 as photocatalysts at pH = 4.



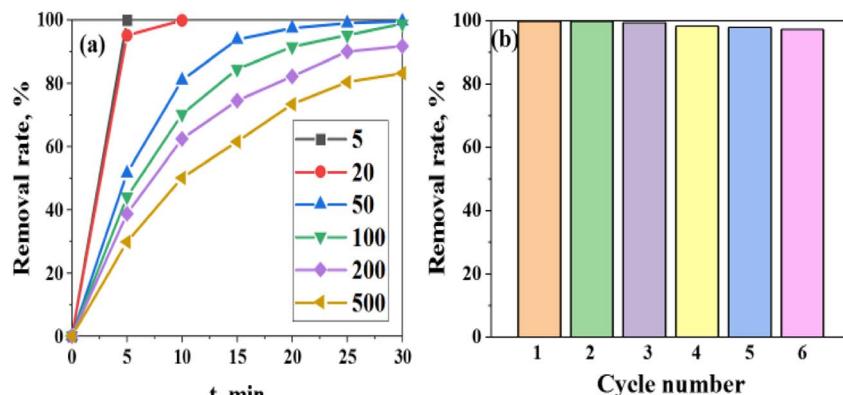


Fig. 10 (a) The degradation efficiency (%) of the (Mo + N)/TiO₂ photocatalyst as a function of the initial MO concentration at pH 4. (b) Reusability studies of the (Mo + N)/TiO₂ photocatalyst.

a starting MO concentration of 50 mg L⁻¹ and pH 4 (Fig. 9(a)). The photocatalytic degradation of most organic pollutants at the liquid–solid interface can be expressed *via* a pseudo-first-order kinetic model equation $\ln(C/C_0) = -Kt$,⁶⁸ where C_0 represents the initial concentration of MO, C represents the concentration of MO at time t and k is the rate constant (min⁻¹).

The MO degradation data obtained using pure TiO₂, MT1, MT2, and MT3 was fitted to the pseudo-first-order model. The correlation constant of the fitted line was calculated to be $R^2 > 0.99$ for the doped TiO₂ samples (MT1, MT2, and MT3) and 0.981 for pure TiO₂, indicating that their photocatalytic degradation process matches the pseudo-first-order kinetic model. A sloped line was generated from the linear correlation between $\ln(C/C_0)$ and t , and it was used to get the values of k , as summarized in Table 1. The apparent reaction rate constant of MT3 was 1.87×10^{-1} min⁻¹ at pH 4, while for undoped TiO₂, a lower reaction rate of 1.83×10^{-2} min⁻¹ was detected, as shown in Fig. 9(b). Therefore, the catalyst (Mo + N) codoped TiO₂ nanocomposite possesses greater photocatalytic activity than pure TiO₂, and the color removal rate considerably increased when (Mo + N)-doped TiO₂ was used as the catalyst.

The effect of initial MO concentration on the photocatalytic behavior of (Mo + N)-doped TiO₂ is important to determine the efficiency of MO degradation.^{69–71} The photocatalytic degradation of MO by (Mo + N)-doped TiO₂ at different initial MO concentrations was investigated under the optimal operating conditions of a single solution system (pH 4, contact time 30 min at 25 °C, 10 mg of the catalyst, and 10 mL of MO solution). Several MO concentrations of 5, 20, 50, 100, 200, and 500 mg L⁻¹ were tested under UV light. The degradation efficiency was reduced as the initial concentration of MO was increased above 100 mg L⁻¹, reaching 83.2% in the case of 500 mg L⁻¹ MO after 30 min of contact time, as shown in Fig. 10(a). Further, the adsorption of MO molecules on the surface of (Mo + N)-doped TiO₂ was realized by increasing the MO concentration. Several layers of adsorbed dye molecules are formed on the catalyst surface at an excessive dose, which inhibits the photoreaction because it hinders the direct contact of the light with the catalyst in order to generate hydroxyl radicals.^{62,69–71}

Photocatalyst stability is one of the most important aspects for industrial processing and reversing the possibility of catalyst reuse.^{62,69} It is considered one of the important indicators of the practical benefits and utility of a photocatalyst.^{62,69} The stability of (Mo + N)-doped TiO₂ as a photocatalyst was examined after the fifth degradation run, as shown in Fig. 10(b). The degradation efficiency decreased from 99.6 to 97.11% after five runs, which demonstrates the higher degradation efficiency and stability of the (Mo + N)/TiO₂ photo-catalyst. The decrease in efficiency is strongly attributed to the loss of adsorption sites owing to the difficulty in removing the degradation by-products from the (Mo + N)/TiO₂ photocatalyst surface, such as sulfur-containing compounds. Therefore, the as-synthesized (Mo + N)-doped TiO₂ catalyst reveals excellent efficiency in the photodegradation of MO dye with minor efficiency loss and great durability.

The photocatalytic activity results obtained in this research were compared with the photocatalytic activity of other titanium dioxide/molybdenum photocatalysts. For example, Kanakaraju *et al.*⁷² synthesized a series of titanium dioxide/molybdenum photocatalysts with different molybdenum concentrations (1–10 wt%) and demonstrated that the best-performing titanium dioxide/molybdenum (3 wt%) photocatalyst yielded a methyl orange photodegradation efficiency of 94.5% within 120 min of UV irradiation. Islam *et al.*⁷³ prepared an Ag/MoO₃/TiO₂ photocatalyst, and its maximum photodegradation efficiency was found to be 95.6% within 300 min of UV irradiation. These results demonstrate that the (Mo + N)-doped TiO₂ catalyst obtained in this research demonstrates better efficiency, as evidenced by its higher photocatalytic activity towards methyl orange (99.6% within 30 min of UV irradiation).

4. Conclusion

A simple one-step hydrothermal process followed by calcination treatment was used to prepare a Mo and N codoped TiO₂ nanocomposite with induced oxygen vacancies. The results reveal that the molybdenum and nitrogen atoms substitute the oxygen atoms in the TiO₂ crystal lattice. The as-prepared nanocomposites revealed excellent photocatalytic activity



toward methyl orange degradation under UV light illumination. The photocatalytic activity of the TiO_2 catalyst was considerably enhanced by the Mo/N co-doping. The photocatalytic activity of doped TiO_2 improved with increasing Mo content from MT1 to MT3. After 30 min of contact time, the MT3 catalyst revealed the highest efficiency of 99.6%. The formation of Ti^{3+} and Mo^{5+} , which work collaboratively with the N dopant to generate oxygen vacancies, plays an important role in visible light absorption and enhancing photocatalytic activity.

Conflicts of interest

There are no conflicts to declare.

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