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Electronic modulation by interfacial bridging between Ir nanoparticle and metal–organic framework to enhance hydrogen evolution†

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In the quest for amplifying the inherent capabilities of noble metals, crafting precise metal–support interfacial bonds stands as a challenging yet fruitful endeavor. Here, we introduce an approach to synthesize exceptionally small metal nanoparticles (NPs), which are securely anchored onto cobalt-based metal–organic framework (Co-MOF) nanosheets through a wet chemical method. The creation of metal–oxygen linkages between these Ir NPs and Co-MOF not only guarantees structural robustness but also ensures a generous exposure of active sites. Importantly, the distinct interface fine-tunes the electronic configuration of the composites by facilitating charge transfer along the Ir–O–Co pathway, consequently boosting the reaction dynamics. As a result, the hydrogen evolution reaction (HER) performance of our pioneering product, Ir@Co-MOF, in alkaline media, rivals that of the benchmark Pt/C catalyst and surpasses other recent noble metal catalysts. Theoretical analysis reveals that the exceptional HER efficiency of Ir@Co-MOF originates from the optimal adsorption energies for water and hydrogen, a consequence of electron re-allocation owing to its unique interfacial bond.

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1. Introduction

In the context of the global transition towards sustainable energy sources, hydrogen energy emerges as a key player due to its clean and efficient characteristics.^{1–3} Specifically, water electrolysis for the hydrogen evolution reaction (HER) has garnered significant attention for its environmental compatibility and high efficiency.^{4,5} However, the effectiveness of the HER is highly dependent on the choice of catalyst. Noble metals such as platinum (Pt), palladium (Pd), and ruthenium (Ru) have traditionally served as the foundation for HER catalysts due to their exceptional catalytic activities.^{6–9} Even so, their extensive application is hindered by high costs, limited availability, and susceptibility to deactivation under specific conditions.

In response to these challenges, researchers have explored various strategies for optimizing the use of noble metals in HER. This includes alloying them with less expensive metals, creating bimetallic or multimetallic compounds, and developing core–shell structures where noble metals form a thin layer over a more abundant core.^{10–15} While these approaches have improved catalytic performance and cost-effectiveness, challenges such as long-term stability and precise control of electronic properties still remain. Metal–Organic Frameworks (MOFs), with their expansive surface area and adjustable pore sizes, have risen as a promising solution to various challenges in the field.^{16–18} Their distinctive structural properties offer an ideal platform for anchoring noble metal nanoparticles, leveraging these features for enhanced performance in various applications.^{19,20} This not only allows for a significant reduction in the amount of noble metal used but also enhances the accessibility of active sites, thereby improving catalytic efficiency. For example, Huang *et al.* successfully synthesized ultra-small Pt nanoparticles (NPs) oriented on MOFs nanosheets by precisely regulating the reduction kinetics of metal ions, which significantly enhanced HER performance.²¹ Furthermore, the structural versatility of MOFs allows for precise adjustment of the electronic environments surrounding the metal sites, which is essential for achieving optimal catalytic performance.^{22,23} Chen and colleagues employed

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MOF-74, known for its structured pore configuration and effective electrical conductivity, as a scaffold to stabilize Pd clusters.²⁴ These clusters were then skillfully transformed into sophisticated HER electrocatalysts (Pd@MOF-74). It is worth noting that the catalysts have a very high precious metal content, with some even reaching 30%. Relative to Pt, Pd and Ru, Ir exhibits superior catalytic performance in HER due to its distinctive chemical and physical attributes, even when used in very small amounts.^{25–27} Ir exhibits higher corrosion resistance and chemical stability, making it particularly suitable for alkaline environments.^{28–30} These characteristics make iridium an ideal candidate for enhancing HER efficiency while reducing the overall cost and quantity of catalyst used.

In summary, the integration of Ir NPs with MOF nanosheets holds great promise for enhancing HER performance. Taking inspiration from this, the immobilization of Ir NPs onto the surface of MOF nanosheets is achieved through a wet chemical strategy. The composite material exhibited remarkable catalytic efficiency as anticipated, confirming the effectiveness of our strategy to merge the catalytic capabilities of iridium with the structural advantages of MOFs. To elaborate, this approach boosts the inherent catalytic activity of Co-MOF and lowers the Gibbs free energy of H⁺ intermediates, thus enhancing the electrocatalytic efficiency of Ir@Co-MOF for the HER. This integration not only underscores the potential of Ir as a formidable catalyst in HER but also highlights the effectiveness of Co-MOF as an ideal platform for nanoparticle dispersion and stability, paving the way for a more sustainable and cost-effective approach to hydrogen production.

2. Results and discussion

The production pathway for Ir@Co-MOF composites is depicted in Fig. 1a. In the initial phase, solvothermal methods are used to grow Co-MOF nanosheet arrays on NF substrates. This is followed by the attachment of Ir NPs onto the MOF nanosheets, utilizing iridium chloride (IrCl₃) as the iridium source, achieved through a wet chemical technique. In this process, *N,N*-dimethylformamide (DMF) and ethanol serve as reducing agents for fabricating Co-MOF nanosheets and converting Ir³⁺ into Ir NPs, respectively, eliminating the requirement for extra reducing agents or stabilizers. Consequently, this approach enhances the electrocatalytic capabilities of the composite, attributed to the highly accessible Ir active sites, unobstructed by any capping agent. The structure of the samples throughout the fabrication stages was examined using a scanning electron microscope (SEM). Initially, the Co-MOF exhibits a sheet-like form (refer to Fig. S1†). After the introduction of Ir NPs, the Co-MOF nanobelt arrays retain their morphology (as shown in Fig. 1b), offering extensive surface areas and rapid mass transport, vital for effective catalysis. As shown in Fig. 1c, transmission electron microscopy (TEM) also showed similar nanosheet morphology. As seen in Fig. 1d, show a uniform distribution of Ir NPs across the Co-MOF surface, each averaging about 1 nm in diameter (Fig. 1e). The

high-resolution TEM (HRTEM) image in Fig. 1f displays a lattice spacing of 0.218 nm, aligning with the (111) plane of Ir.³¹ Energy-dispersive X-ray spectroscopy (EDX) mappings, depicted in Fig. 1g, confirm the even spread of Co, Ir, and O elements over the entirety of the Co-MOF surfaces.

The X-ray diffraction (XRD) pattern of Ir@Co-MOF, shown in Fig. S2,† closely resembles that of the unmodified Co-MOF, which shares an isostructural relationship with Co-MOF (CCDC no. 985792).³² The absence of distinct Ir NPs peaks in the XRD pattern can likely be attributed to their low concentration and diminutive size, a hypothesis is supported by an inductively coupled plasma analysis indicating an Ir content of approximately 6.27 wt%. To delve deeper into the interface interactions between Ir and Co-MOF, X-ray photoelectron spectroscopy (XPS) was employed. Fig. S3† showcases a side-by-side comparison of the XPS spectra for Co-MOF, both in its original form and after Ir integration. In addition to the standard peaks for C, O, and Co, two new, distinct peak range from 300 to 500 eV emerges, signifying the presence of Ir in the resulting composite.³³ The high-resolution XPS spectrum of Ir 4f, depicted in Fig. 2a, reveals peaks at 60.8 and 63.8 eV, corresponding to the Ir 4f_{7/2} and Ir 4f_{5/2} states of elemental Ir (0), respectively. Moreover, the appearance of peaks at 62.0 and 65.4 eV in the XPS spectrum indicates the formation of Ir–O bonds within the composite.¹⁹ The high-resolution Co 2p spectrum of Co-MOF, as shown in Fig. 2b, reveals peaks for Co 2p_{1/2} and Co 2p_{3/2} at 797.2 and 781.1 eV, respectively, along with their satellite peaks, confirming the presence of Co²⁺. Notably, the incorporation of Ir NPs leads to a significant negative shift in these Co 2p_{1/2} and Co 2p_{3/2} peaks, which is indicative of a substantial charge transfer from Ir NPs to Co-MOF, further substantiating the interaction between the two components. This data suggests that the electronic structure of the Ir and Co sites is modifiable through interfacial coupling interactions, significantly influencing their catalytic activity in electrochemical reactions. In Fig. S4,† the O 1s spectra of Ir@Co-MOF are meticulously analyzed, showcasing three distinct peaks at energy levels of 531.2, 531.7, and 532.9 eV.²⁵ These peaks are respectively indicative of Co–O, O=C=O, and adsorbed H₂O species. Additionally, a notable peak at 530.7 eV in the Ir@Co-MOF spectrum is observed, signifying the formation of interfacial Ir–O–Co bonds. Moreover, the high-resolution C 1s spectra, illustrated in Fig. S5,† display three distinct peaks at 284.2, 284.6, and 288.0 eV. These are respectively associated with the C=C, C–O, and C=O bonds.³⁴

To enhance understanding of the local electronic state and the emergence of Ir–O–Co bridges, X-ray absorption fine structure (XAFS) analysis was utilized. Regarding the Ir LIII-edge XANES spectra shown in Fig. 2c, the sequence of increasing white line intensity is as follows: Ir foil, Ir@Co-MOF, and then IrO₂, which reflects the ascending order of the valence states of the Ir species. The marginally elevated valence state of Ir@Co-MOF compared to Ir foil further substantiates the presence of oxides within the Ir@Co-MOF and corroborates the reduction in electron density at the Ir sites. The Co K-edge XANES spectra, as depicted in Fig. 2d, reveal a shift in the

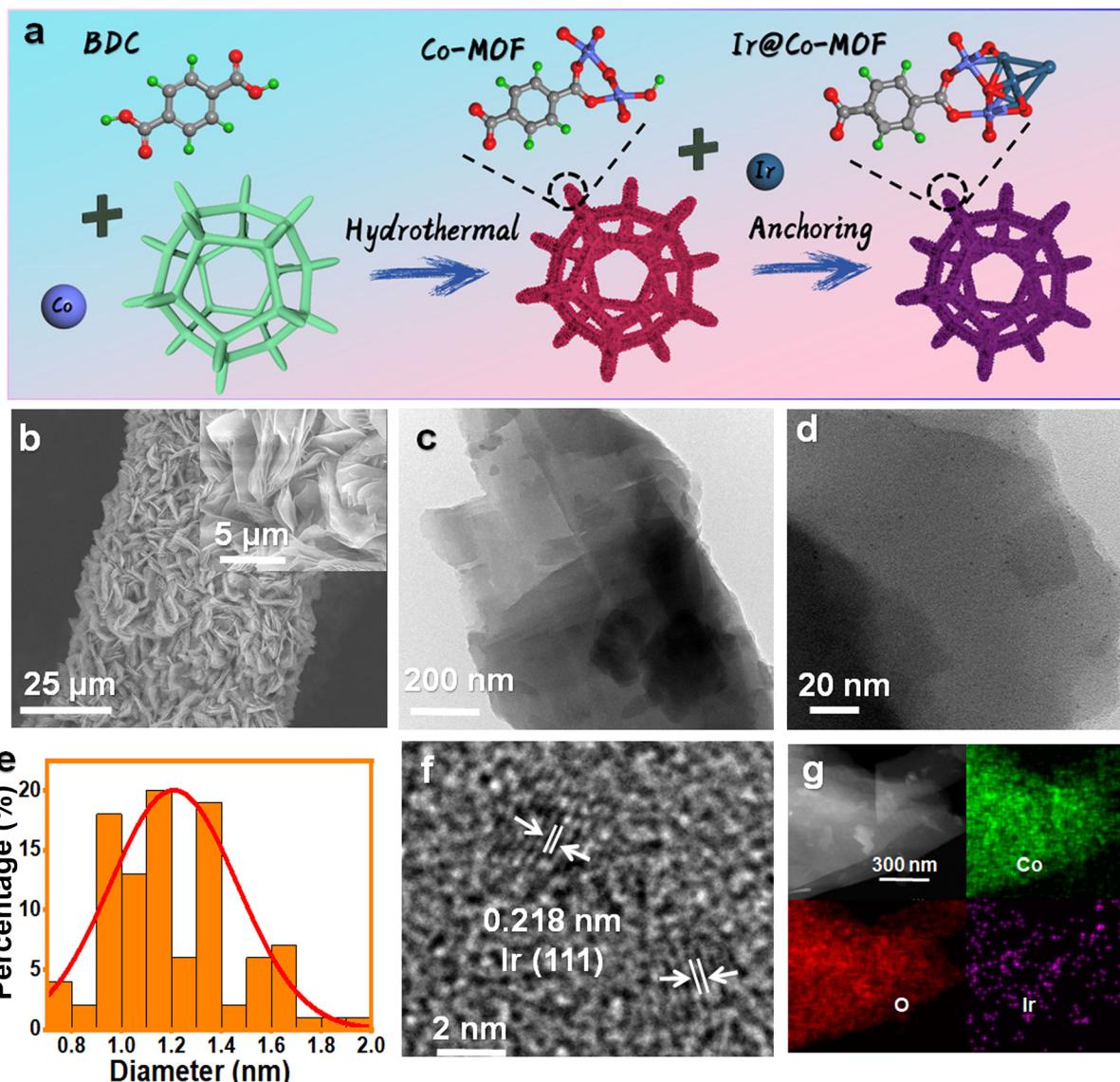


Fig. 1 (a) Schematic illustration of the synthesis of Ir@Co-MOF. (b) images of Ir@Co-MOF. (c and d) TEM images at different magnifications of Ir@Co-MOF. (e) The size distribution of Ir nanoparticles. (f) HRTEM and (g) EDS mappings of Ir@Co-MOF.

absorption edge of Ir@Co-MOF to a lower energy than that of Co-MOF. This shift suggests an increase in electron density around Co sites due to the immobilization of Ir NPs on Co-MOF.³⁵ Furthermore, the pronounced amplitude variation in the oscillation of the Co K-edge extended X-ray absorption fine structure (EXAFS) function between Ir@Co-MOF and Co-MOF, as shown in Fig. S6,[†] implies distinct local atomic configurations. Evidently, the XANES findings are entirely consistent with the XPS data, demonstrating that substantial electron transfer occurs from Ir to Co-MOF. Fig. 2e presents the Ir LIII-edge EXAFS spectra, illustrating that, relative to Ir foil, the presence of shorter Ir-Co bonds in the Ir@Co-MOF results in a shorter Ir-Ir bond length and a decreased coordination number for Ir in the Ir-Ir bonds. Furthermore, the coordination number of Ir within the Ir-O bond rises after forming

the composite, further confirming the establishment of the Ir-O bond at the heterostructure interface. The EXAFS spectra of Ir@Co-MOF, illustrated in Fig. 2f, indicate that the primary peak corresponds to Co-O single scattering paths, while weaker peaks are related to higher shell interactions involving Co-O-Co/Ir.^{36–38} Notably, the increased intensity of the Co-O-Co/Ir bond peak in Ir@Co-MOF compared to Co-MOF corroborates the formation of Ir-O-Co bridges.³⁹ This bond's shift suggests robust interaction at the Ir@Co-MOF heterostructure interface, a finding supported by wavelet transform spectroscopic analysis of the Co K-edge EXAFS (Fig. 2g). These combined results point to the formation and electron transfer mechanism in Ir@Co-MOF featuring interfacial Ir-O-Co bridging. Consequently, the joint analysis of XPS and XANES confirms the presence of this bridging in Ir@Co-MOF, which

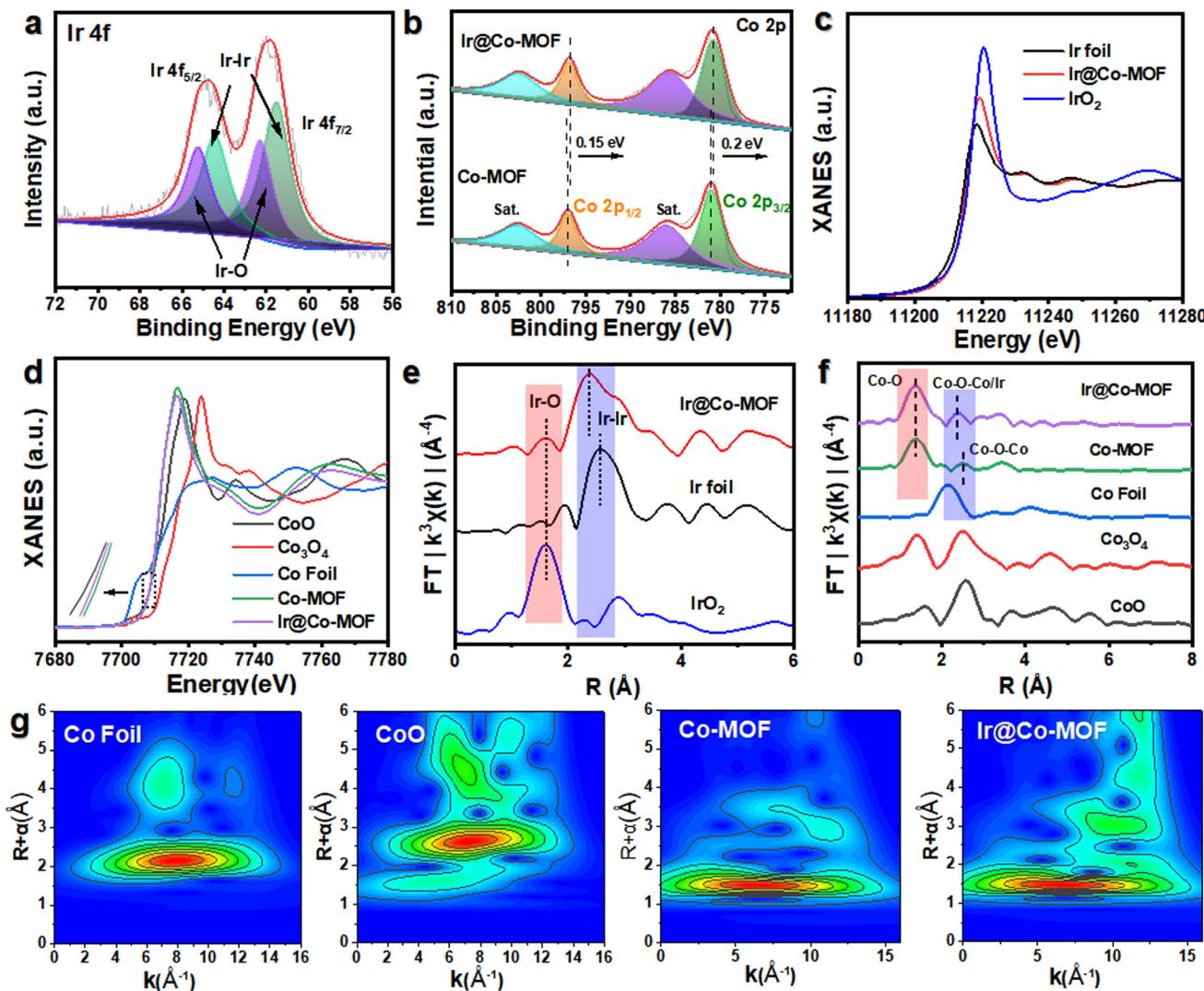


Fig. 2 XPS spectra of prepare samples. (a) Ir 4f, (b) Co 2p, of Ir@Co-MOF and Co-MOF. (c and d) The normalized XANES spectra (e and f) FT-EXAFS spectra and (g) Wavelet transform for k^3 -weighted EXAFS signal of Ir foil, IrO₂, Co Foil, CoO, Co-MOF and Ir@Co-MOF, respectively.

alters the local electronic structure of the catalyst. This alteration potentially improves the adsorption/desorption dynamics of reaction intermediates, thereby enhancing the material's electrocatalytic efficiency.

To evaluate the catalytic prowess of Ir@Co-MOF, we conducted tests on its HER activity within a three-electrode cell framework, using 1 M KOH as the test medium. For a comparison, we also examined Co-MOF, bare NF (nickel foam), and a commercially available Pt/C coated on NF (Pt/C) under identical experimental conditions. The linear sweep voltammetry (LSV) curves reveal a noteworthy performance of the Ir@Co-MOF electrocatalyst in HER. The results, as presented in Fig. 3a, show remarkable efficiency, with the system reaching a current density of 10 mA cm^{-2} at an exceptionally low overpotential of just 29 mV. This discovery of this is further supported by comparing the overpotentials required at different current densities, specifically at 10 mA cm^{-2} , 50 mA cm^{-2} and

100 mA cm^{-2} (Fig. 3b). The Ir@Co-MOF exhibited low overpotentials at these current densities, measuring 29, 71, and 99 mV, respectively. These values are significantly lower than those for Co-MOF, which were 197, 246, and 286 mV, respectively, and are on par with commercial Pt/C catalysts, which showed overpotentials of 24, 35, and 50 mV. The Tafel value for Ir@Co-MOF (87.6 mV dec^{-1}) is lower compared to Co-MOF ($123.6 \text{ mV dec}^{-1}$) (Fig. 3c), indicating Ir@Co-MOF demonstrates faster H₂ generation kinetic. The electrochemical impedance spectroscopy (EIS) results, including the Nyquist plots of Ir@Co-MOF and Co-MOF, are employed to analyze the efficiency of charge transfer between the Ir nanoclusters and the supporting materials.⁴⁰ Fig. 3d displays these Nyquist plots, where a notably smaller diameter of impedance arc (DIA) for the Ir@Co-MOF modified electrode, compared to commercial Co-MOF, indicates enhanced interfacial charge transport. The impedance spectra were fitted to an equivalent

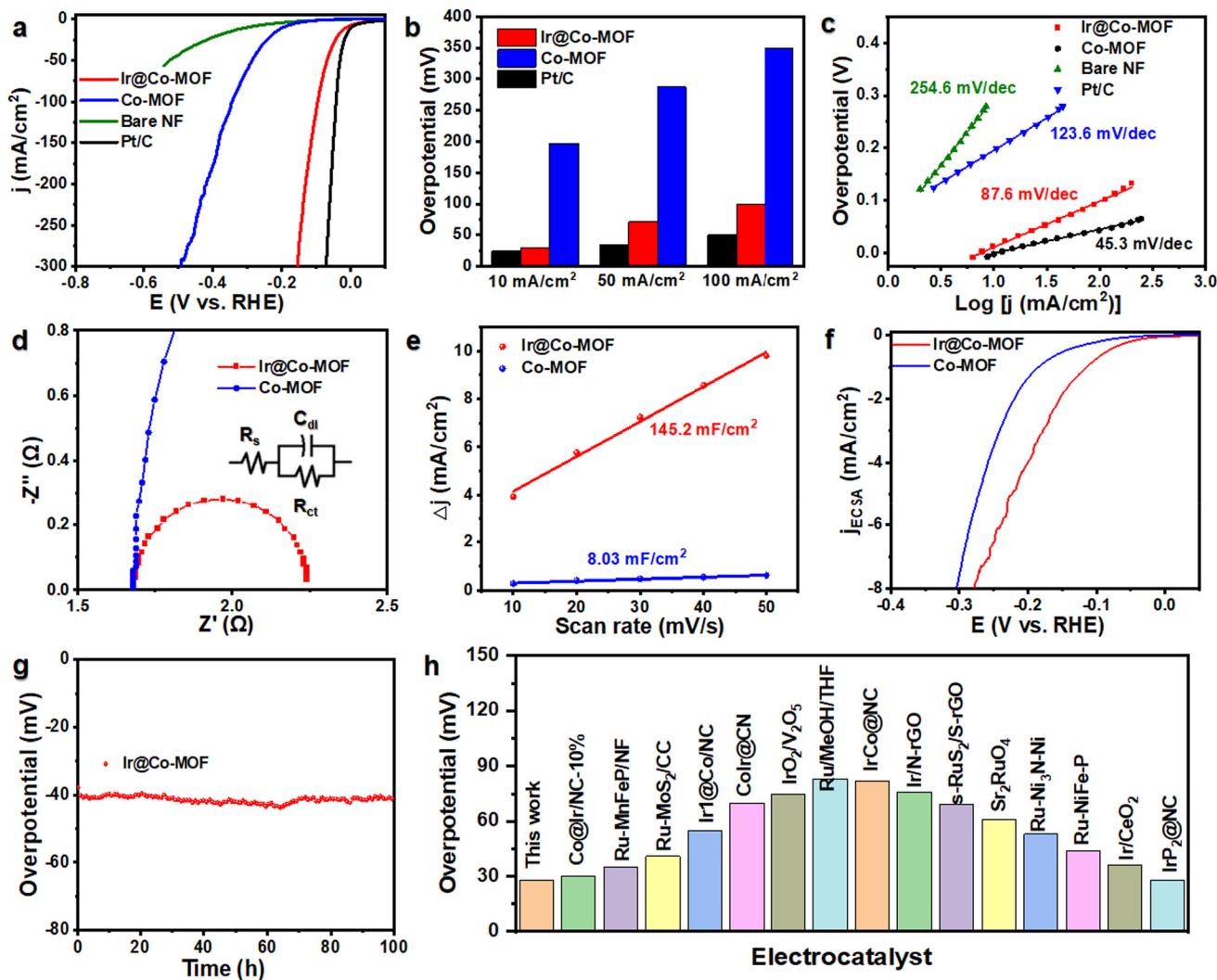


Fig. 3 (a) HER polarization curves for Ir@Co-MOF, Co-MOF, Pt/C, and Bare NF. (b and c) Corresponding overpotentials at 10 mA cm^{-2} , 50 mA cm^{-2} , 100 mA cm^{-2} and Tafel plots. (d) Nyquist plots of different samples. (e) Double-layer capacity C_{dl} of Ir@CoMOF and Co-MOF. (f) LSV curve after ECSA normalization. (g) Chronoamperometry curves. (h) HER activity comparison graph of the overpotentials at 10 mA cm^{-2} for the reported Pt-based.

circuit model denoted as $R(QR)$, where R_s symbolizes the electrolyte resistance, R_{ct} denotes the charge-transfer resistance, and Q_s represents the constant phase elements (CPE).⁴¹ To delve deeper into the intrinsic activity, we assessed the electrochemical active surface area (ECSA) by calculating the double-layer capacitance (C_{dl}) at various scan rates during cyclic voltammetry (CV) measurements. These measurements were conducted within voltage ranges that excluded redox processes, as shown in Fig. S7.† In comparison to the pristine Co-MOF with a C_{dl} value of 8.03 mF cm^{-2} , the Ir@Co-MOF hybrids exhibit a substantial increase in C_{dl} , measuring 145.2 mF cm^{-2} , as illustrated in Fig. 3e. This increase indicates that the introduction of Ir NPs provides a significantly larger number of active sites. To eliminate the influence of ECSA on performance, we have calculated and compared the normalized current density, as depicted in Fig. 3f. Furthermore, the Ir@Co-MOF exhibits the highest ECSA normalized current density for the HER, under-

scoring its superior intrinsic activity. To further assess the inherent catalytic capabilities of the catalysts, we computed the turnover frequencies (TOF) for each, details of which can be found in the ESI.† The TOF values for the Ir@Co-MOF catalysts were significantly higher than those observed for Co-MOF. Specifically, at an overpotential of 100 mV , the TOF value for each active site on Ir@Co-MOF was measured at 0.07 s^{-1} , which is 5.4 times higher than the TOF of 0.013 s^{-1} recorded for Co-MOF. Additionally, the amount of hydrogen produced was quantified using the water displacement method (refer to Fig. S8†). The volume of H_2 measured experimentally aligns closely with the theoretical predictions. Consequently, we were able to achieve a faradaic efficiency of 99% with the Ir@Co-MOF catalyst.

In addition to its electrocatalytic activity, the Ir@Co-MOF demonstrates excellent durability, with no significant decay observed during a chronoamperometry measurement con-

ducted over 100 hours at 10 mA cm^{-2} , as depicted in Fig. 3g. The exceptional durability of the Ir@Co-MOF composite can be attributed to its unique heterostructure. In this structure, the Co-MOF nanosheets play a crucial role in preventing the aggregation of Ir NPs, thanks to the strong interface interaction between them. Furthermore, it's worth noting that Ir@Co-MOF exhibits outstanding performance in alkaline HER with a low overpotential, surpassing the performance of most noble-metal-based electrocatalysts that have been reported to date (Fig. 3h and Table S1†). The SEM and XRD characterization of Ir@Co-MOF after HER reveals that their nanosheet morphologies remain almost unchanged (Fig. S9 and S10†). Furthermore, XPS analyses of the Ir@Co-MOF catalyst, as shown in Fig. S11,† confirm the stability of the Ir-O bond, which remains intact even after the electrochemical testing. These results collectively highlight the remarkable electrochemical performance and endurance of the Ir@Co-MOF catalyst in HER applications.

To investigate the impact of the synthesis method on performance, we incorporated IrCl_3 during the MOF synthesis. The results indicated that while the direct integration of Ir ions in a one-step process preserves the crystallinity of the MOF, it significantly compromises the structural integrity of the MOF nanosheets, adversely affecting their catalytic performance (Fig. S12†). Moreover, IrCl_3 content was varied to produce $\text{Ir}_{1.32} \text{ wt\%} @ \text{Co-MOF}$ and $\text{Ir}_{13.53} \text{ wt\%} @ \text{Co-MOF}$ materials with Ir contents of 1.32 wt% and 13.53 wt%, respectively. The XRD pattern, as depicted in Fig. S13,† indicates that the incorporation of Ir into Co-MOF has a negligible impact on its crystalline structure. Despite this, the presence of Ir markedly enhances the electrocatalytic performance. The polarization

curves in Fig. S14† clearly illustrate that samples containing 6.27 wt% of Ir display the most effective HER performance. Specifically, the $\text{Ir}_{6.27} \text{ wt\%} @ \text{Co-MOF}$ sample distinguishes itself by exhibiting the lowest charge transfer resistance and the most advantageous Tafel slope among all tested samples, as evidenced in Fig. S15.† Additional electrochemical tests, detailed in Fig. S16,† further confirm the exceptional HER activity of Ir@Co-MOF. This enhanced activity is attributed to its minimal charge transfer resistance and maximal ECSA. Collectively, these findings indicate that a 6.27 wt% concentration of Ir optimally improves the structural stability, catalytic activity, and interfacial synergistic effects in Co-MOF. In pursuit of further validating the universality of our synthetic methodology, we extended our investigations to include additional metal-organic frameworks (MOFs), those derived from biphenyldicarboxylate (BPDC) and benzene-1,3,5-tricarboxylate (BTC). XRD analysis confirmed the successful synthesis of these MOFs and their similar nanosheet structures (Fig. S17†). Subsequent evaluations focused on the HER performance of these synthesized MOFs to assess their catalytic efficacy. Among the various MOFs tested, the framework synthesized using benzene-1,4-dicarboxylate (BDC) exhibited superior performance (Fig. S18†). This result not only emphasizes the versatility of our synthetic approach but also highlights the potential for optimizing electrocatalytic performance through strategic selection of MOF precursors.

To elucidate the mechanism behind the high activity and outstanding kinetics of Ir@Co-MOF, DFT calculations were performed. In Fig. 4a and b, you can observe the structure model of Ir@Co-MOF. As depicted in Fig. 4c, it is evident that there is a rearrangement of electrons at the interface, leading

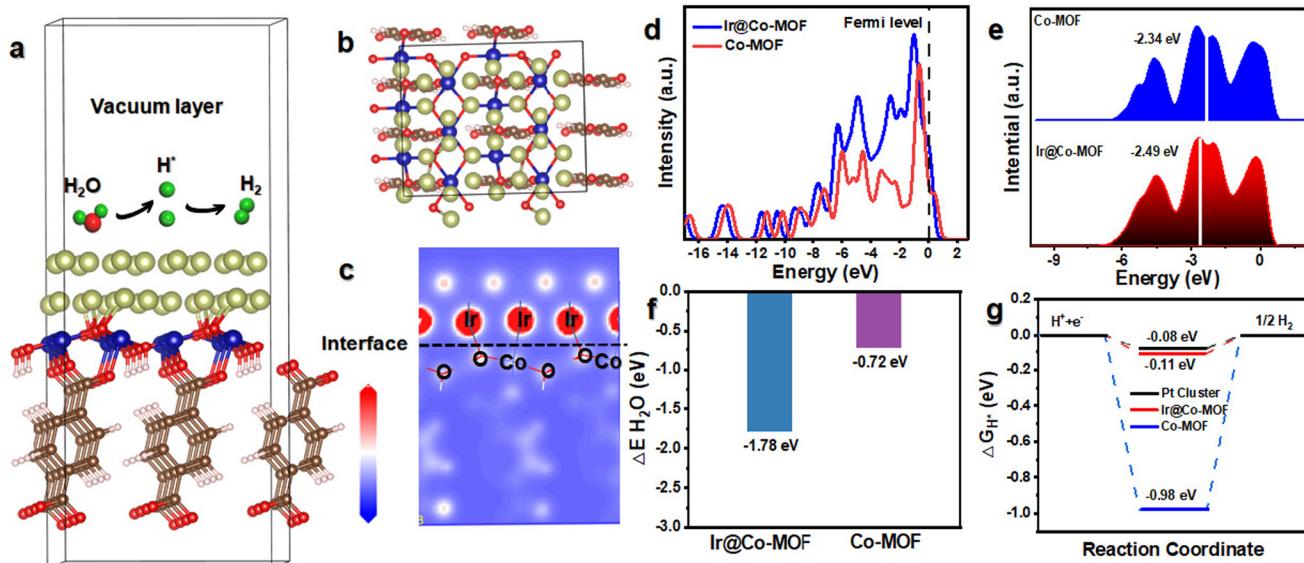


Fig. 4 (a and b) HER reaction mechanism on the Ir@Co-MOF model (green, light blue, red, and dark blue represent hydrogen, cobalt, oxygen, and iridium atoms, respectively). (c) Differential charge density of the Ir@Co-MOF model. (d) DOS of Ir@Co-MOF and Co-MOF, where the red line represents the Fermi level. (e) Partial electronic density of states of the Co d orbital in Ir@Co-MOF and Co-MOF. (f) H_2O adsorption energy for the three systems. (g) Calculated free-energy diagram of H adsorption.

to a loss of electrons at the Ir site and an increase of electrons at the Co site. This observation aligns with the XPS and XANES analysis results and provides further evidence of the strong electronic interaction between the Ir NPs and the Co-MOF substrate. The analysis of the total density of states (DOS) for Ir@Co-MOF indicates an increased electron density near the Fermi level, especially when contrasted with that of Co-MOF. The incorporation of Ir NPs into the structure, as depicted in Fig. 4d, implies an enhancement in electrical conductivity and electron transfer efficiency during the HER. Additionally, this electron transfer adeptly modulates the d-band center of Ir. As illustrated in Fig. 4e, this modulation involves shifting the d-band center towards a lower energy level at the interface, which in turn facilitates the adsorption of hydrogen onto the surface.^{42,43} Such a shift is instrumental in boosting the catalytic performance of the system. The calculated adsorption energies of H₂O and H* on the surfaces of Ir@Co-MOF and Co-MOF were further analyzed. Comprehending the adsorption process of H₂O molecules on the catalyst surface is essential for facilitating water dissociation in alkaline environments during the HER.^{44–46} The analysis of binding energies for H₂O on both pure Co-MOF and the enhanced Ir@Co-MOF reveals that the adsorption of H₂O is more thermodynamically favorable on Ir@Co-MOF, as depicted in Fig. 4f. This observation implies that H₂O molecules can be effectively adsorbed and activated on Ir@Co-MOF, thereby enhancing the HER process. Moreover, the adsorption free energy of hydrogen atoms (H*) denoted as ΔG_{H^*} is pivotal in determining the efficiency of HER.^{47,48} Values of ΔG_{H^*} nearing zero suggest an optimal balance between hydrogen adsorption and H₂ desorption. Illustrated in Fig. 4g, the ΔG_{H^*} for Ir@Co-MOF is -0.11 eV, significantly closer to the ideal thermoneutral value than that of pure Co-MOF (-0.98 eV). This indicates reduced barriers in the adsorption–desorption cycle, leading to an elevated HER performance for Ir@Co-MOF. These results demonstrate that the electronic structure modification, achieved through the formation of Ir–O–Co bonds, synergistically optimizes the free energies associated with hydrogen and water adsorption, contributing to the superior HER activity. This optimization facilitates the HER process in alkaline solutions, leading to the exceptional performance of Ir@Co-MOF as an electrocatalyst.

3. Conclusion

In summary, a simple wet chemical strategy has been successfully employed to synthesize Ir NPs anchored on Co-MOF nanosheet arrays (Ir@Co-MOF), resulting in high-performance electrocatalysts for the HER in alkaline solutions. The Co-MOF nanosheets act as a durable support for the catalyst, effectively preventing the agglomeration of Ir NPs. This stability ensures a vast exposure of active surface sites and facilitates efficient electron transfer across interfaces *via* the Ir–O–Co bonds. As a result, the finely tuned Ir@Co-MOF electrocatalyst showcases exceptional HER activity. It achieves a notable current density

of 10 mA cm^{-2} at a remarkably low overpotential of just 29 mV in an alkaline setting. Additionally, this catalyst maintains its impressive performance with remarkable long-term stability. This promising electrocatalyst holds significant potential for various applications in hydrogen generation and energy conversion technologies. Theoretical calculations have provided valuable insights into this work, revealing that the charge relocation induced by interfacial bonds can enhance the kinetics of H₂O adsorption and improve the kinetics of HER by reducing the barriers for hydrogen adsorption. This research not only improves the catalytic performance of precious metals through electronic structure modulation but also introduces a novel approach to the synthesis of efficient HER catalysts utilizing MOFs as a platform.

Author contributions

Kaipeng Cheng: conceptualization, methodology, writing – original draft. Yinghuan Xu: methodology. Shenmeng Zhang: data curation. Heng Zhang: investigation, software, formal analysis. Yingying Xue: formal analysis. Xiaohui Huang: validation. Jianbo Zhao: data curation. Pengcheng Zhang: investigation, Shuang Ma: formal analysis, data curation. Zhiqiang Hou: writing – review & editing, funding acquisition.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper. Data availability Data will be made available on request.

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