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Phenoxy imine NOON complexes of heavy alkaline earth ions for the ring-opening polymerisation of cyclic esters†

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The polymerisation of lactide monomers by heavier alkaline earth complexes supported by a phenoxy-imine NOON ligand, $^{\text{Dipp}}\text{LM}(\text{thf})_x$ ($^{\text{Dipp}}\text{L} = [2,7\text{-}(C=\text{NDipp})\text{-}1,8\text{-}O\text{-}C_{10}\text{H}_4]^2\text{-}}$ where Dipp = $2,6\text{-}^{\text{i}}\text{Pr}\text{-}C_6\text{H}_3$; M = Ca (1), Sr (2) and Ba (3)) is described. Complexes 1-3 all proved active initiators for the ROP of LA converting over 500 eq. at ambient temperatures. The dependency on monomer concentration was found to vary depending on the alkaline-earth metal used; calcium-based initiator 1 exhibited first-order dependency with respect to monomer in all cases, while there was a second order dependancy for the heavier strontium analogues. Addition of benzyl alcohol as a co-initiator resulted in an ~8-fold and ~6-fold increase in the observed rate for 1 and 2 respectively. In these systems, presence of the expected $-\text{OCH}_2\text{Ph}/-\text{OH}$ end groups, formed via the activated monomer pathway, was confirmed by ^1H and ^{13}C (^1H) NMR spectra of the isolated polymers. Further investigations with complex 2 suggested that without addition of co-initiator, the ROP could proceed via an anionic mechanism in which the initiating step would be monomer deprotonation. In all cases, poor polymerisation control was observed. GPC analysis of the polymers indicated the presence of different initiating species, which may be due to the complex speciation of complexes 1-3 or the formation of a mixture of linear and cyclic PLA.

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Introduction

Polylactide (PLA) is a biodegradable, aliphatic polyester that can be derived from renewable resources (corn, beets) thus making it a sustainable alternative to polyolefins. It also has relatively improved end of life options which include hydrolysis and subsequent enzyme-catalysed degradation into $\rm CO_2$ and $\rm H_2O$. 2,3

Recent life cycle analyses suggest that the formation of PLA results in reductions in both GHG emissions (~40%) and non-renewable energy usage (~25%) when compared to petrochemically derived plastics, ⁴⁻⁶ though the true life cycle of all such polymers should be carefully considered in the quest for a more sustainable future and a circular economy.⁷

Polylactide is one of the most widely studied "green polymers" due to its already proven versatile applications in packaging, films, pharmaceuticals and even biomedical engineering (it has been utilised in 3D printing techniques to fabricate tissue scaffolds). ^{28,10} As a result, its global production capacity has almost doubled, and it now accounts for 20.7% of all bio-

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plastics.¹¹ Market analysis reports suggest that this will continue to increase with an estimated compound annual growth rate of 21.4% from 2023 to 2030.⁹

To date, the most successful route for the production of high molecular weight PLA is the ring-opening polymerisation (ROP) of lactide (LA). This method also facilitates enhanced control over other polymer macromolecular properties including polydispersity, stereoselectivity and nature of end-groups.¹²

Previously reported initiators for this process include (single-site) metal complexes, $^{13-23}$ organic species $^{24-26}$ and even enzymes. $^{27-29}$

The two most common mechanisms reported for the ROP of LA are the coordination-insertion and activated monomer pathways. ^{15,30} The majority of neutral (single-site) metalmediated polymerisations are thought to operate *via* coordination-insertion whereas the activated monomer route is more prevalent for cationic and organocatalytic systems. ^{24,31} The anionic mechanism for the ROP of LA is less widely studied. It typically involves ionic, nucleophilic catalysts such as alkali metal alkyls/alkoxides. ^{10,32-37} It is proposed to proceed *via* one of two initiation steps: monomer deprotonation or direct nucleophilic attack. ³² The active anionic chain ends, produced in both cases, are commonly observed to cause unwanted racemisation and/or transesterification reactions. As a result, poor polymerisation control, as indicated by

broad molecular weight distributions, is a characteristic feature of anionic ROP.33

In the context of lactide polymerisation, there have been numerous reports of well-defined bimetallic M²⁺ systems with N/O-donating ligands; the majority are magnesium-based and incorporate highly active amide/alkoxide initiating groups. 38-40 Investigations of analogous heavier bis(group 2) systems are much less common due to their propensity to redistribute in solution.41 There are however, examples of monometallic Ca, Sr and Ba LA ROP catalysts; the coordination sphere of these complexes is typically filled with donating solvent molecules as opposed to dedicated initiating groups. 13,42-44 This precludes polymerisation activity via the typical coordination-insertion and, in the absence of a co-initiator, activated monomer pathways.

Bimetallic calcium and strontium 2,4-di-tert-butyl phenolates (DBP) have been studied, by Mountford et al., as initiators for the ring-opening polymerisation of lactide. 41 The tethered calcium system (Chart 1a) was capable of converting 89% of 100 eq. of rac-LA in 150 minutes. 41 During the same time, 86% conversion was achieved with the dinuclear strontium catalyst (Chart 1b).⁴¹

A series of group two iminopyrrolyl compounds, with no dedicated initiating groups, have been reported as efficient catalysts for the ring-opening polymerisation of the cyclic ester ε - caprolactone (Chart 1c).42,43 The calcium analogue showed comparable activity to that of the magnesium variants. 43 In contrast, the strontium and barium systems showed improved reactivity (600 eq. were converted within 5-10 minutes at 25 °C) without compromising polymerisation control (M_w/M_p) = D < 1.59). At The authors report that this is due to the larger ionic radii of the Sr2+ and Ba2+ cations increasing the availability of space for initiator-monomer interactions. 43,44

A similar trend in polymerisation activity (Ca: 0.137 < Sr: $0.149 < Ba: 0.839 \text{ M}^{-1} \text{ h}^{-1}$; E = S; D < 1.60) was observed in related chalcogen containing systems (Chart 1d).44

We have recently reported the synthesis of alkaline earth complexes supported by a phenoxy-imine NOON ligand, DippL $([2,7-(C=NDipp)-1,8-O-C_{10}H_4]^{2-}$ where Dipp = $2,6^{-i}Pr-C_6H_3$) and found the oxophilic M2+ cations to preferably bind to the central [O-,O-] coordination site. 45 As such, a well-defined dimeric magnesium complex [DippLMg(thf)]2 was isolated. For calcium, strontium and barium, equilibrium mixtures of monomeric and dimeric species were formed; $^{\text{Dipp}}LM(\text{thf})_x =$ $[^{\text{Dipp}}LM(thf)_x]_2$ (M = Ca; $1(thf)_2/[1(thf)_2]_2$, M = Sr; $2(thf)/[2]_2$ $(thf)_{2}$ and M = Ba; $3(thf)/[3(thf)]_{2}$; these equilibrium mixtures will be referred to as 1 (Ca), 2 (Sr) and 3 (Ba) hereafter for simplicity (Chart 1e).

Chart 1 (a) and (b) Bimetallic calcium and strontium phenolates, reported by Mountford et al., for the ring-opening polymerisation of lactide; 41 (c) and (d) previously reported monomeric, heavier alkaline-earth initiators for the ring-opening polymerisation of lactide, 42-44 and (e) this work, DippLM $(thf)_x = [^{Dipp}LM(thf)_x]_2 (M = Ca; 1(thf)_2/[1(thf)_2]_2, M = Sr; 2(thf)/[2(thf)]_2 and M = Ba; 3(thf)/[3(thf)]_2).$

In light of the paucity of heavier alkaline earth complexes utilised for the ring-opening polymerisation of lactide monomers; we report studies into the capabilities of heavier alkaline earth complexes 1–3 to polymerise lactide monomers, exploring the kinetics of the reaction for complex 2 in depth.

Results and discussion

Preliminary lactide polymerisation studies using $^{Dipp}LM(thf)_x$; M = Ca (1), Sr (2) and Ba (3)

The speciation of complexes 1–3 is not well-defined; these complexes are dimeric in the solid-state but exist as an equilibrium mixture of monomer and dimer in solution. In order to collect consistently comparable results, the stoichiometries in the following polymerisation studies were calculated with respect to the molecular weight of the monomeric, mono-

Table 1 ROP of L-LA using **1–3** with $[L-LA]_0$: $[M]_0 = 500:1$ in benzene-d₆ at 40 °C^a

M	t (h)	Conv. ^b (%)	$k_{\text{obs}}^{c} (h^{-1} \text{ or } M^{-1} h^{-1})$	$R^{2 c}$	$\frac{M_{ m n}}{({ m GPC})^e}$	$M_{\rm n}$ (calcd) f	Đ
1	118	88	0.018 ± 0.02^{c}		55 300	60 705	1.59
2	5	66	0.82 ± 0.2^d	0.971	34400	54784	1.69
3	1	94	_	_	40 600	67 590	1.55

^a Conditions: [L-LA]₀: [M]₀ = 500:1, [L-LA]₀ = 0.5 M in 0.6 mL benzene- d_6 at 40 °C. ^b Measured by ¹H NMR spectroscopic analyses; an average of polymerisation runs. ^c First-order rate constant (k_{obs} ; h⁻¹) and R^2 were obtained from plots of $\ln([\text{L-LA}]_0/[\text{L-LA}]_t)$ vs. time. ^d Second-order rate constant (k_{obs} ; M⁻¹ h⁻¹) and R^2 were obtained from plots of $1/[\text{L-LA}]_t$ vs. time. ^e Determined by GPC in the against PS standards using the appropriate Mark-Houwink corrections. ⁴⁶ f Calculated M_n for PLA synthesised = (conv.(%) × [L-LA]₀/[M]₀) × 144.13.

metallic alkaline-earth systems. For simplicity, the initiators will be referred to as 1 (Ca), 2 (Sr) and 3 (Ba) throughout.

Initial testing of complexes **1**, **2** and **3** as initiators in the ring-opening polymerisation of lactide showed that all three systems were capable of polymerising 500 eq. of L-LA at ambient temperatures (Fig. S1† and Table 1).

The activity of the catalysts followed the trend: Ba > Sr > Ca with 94%, 29% and 4% conversion after 1 h for 3, 2 and 1 respectively. Similar patterns have been widely reported and are rationalised by the larger cations being less sterically congested and having more labile bonding – both of which result in lower kinetic barriers towards monomer coordination/insertion. The reaction catalysed by 3 was considerably faster, and accurate acquisition of kinetic data was therefore not obtained (even at higher [L-LA]₀/[Ba]₀ ratios).

In contrast, plots of $\ln([\text{I-LA}]_o/[\text{I-LA}]_t) vs.$ time for the Ca (1) and Sr (2) initiated polymerisations, enabled experimental rate constants to be determined (Fig. S1†). For 1, the first-order rate constant was calculated to be 0.018 h⁻¹; this value aligns with others reported for heavier alkaline-earth bis(diphenyl-phosphinothioic/selenoic) amine catalysts (k_{obs} range from 0.15–0.43 h⁻¹ for Ca; 0.15–1.21 h⁻¹ for Sr and 0.906–1.54 h⁻¹ for Ba for $[\text{LA}]_o/[\mathbf{M}]_o = 200).^{44,48}$ The kinetic data for the analogous strontium system had a much poorer correlation with the first-order linear fit ($R^2 = 0.971$). This suggested that the polymerisation may be second-order with respect to monomer concentration and thus the true rate constant is 0.82 M⁻¹ h⁻¹ in these conditions. The second-order nature of the Sr-catalysed polymerisations was confirmed, via plots of $1/[\text{LA}]_t vs.$ time, in further experiments (Fig. 1).

The polymerisations using 1, 2 and 3 all proceeded with moderate control; this was reflected in broad molecular weight distributions which were bimodal for 2 and 3 (Fig. S2 \dagger), and indicated by intermediate dispersity values (D = 1.55-1.69; Table 1). These values are comparable to those reported for

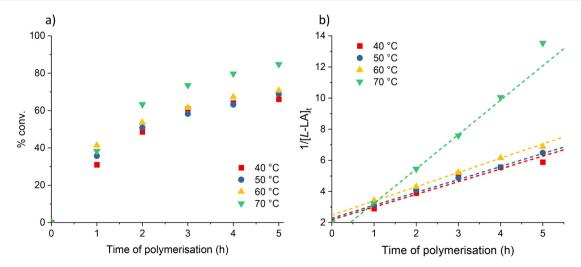


Fig. 1 (a) Plots of percentage conversion vs. time. (b) Plots of $1/[L-LA]_t$ vs. time for L-LA polymerisation using 2. Red squares: 40 °C ($k_{obs} = 0.82 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.963$); blue circles: 50 °C ($k_{obs} = 0.83 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.993$); yellow up triangles: 60 °C ($k_{obs} = 0.91 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.992$); green down triangles: 70 °C ($k_{obs} = 2.20 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.999$). Conditions: $[L-LA]_0 : [Sr]_0 = 500 : 1$, $[L-LA]_0 = 0.5 \text{ M}$ in 0.6 mL benzene- d_6 at stated temperature.

other heavier alkaline-earth, monometallic initiators: D =1.01-1.57. 44,48 The bimodal nature of the molecular weight distributions can be rationalised by the presence of both linear and cyclic PLA and the presence of different initiating species as a consequence of catalyst speciation (vide infra). 41,49,50 The experimental molecular weights however, are slightly lower than the previously documented range for the polymerisation eq. using Ca, Sr and (67 000-74 100 gmol⁻¹). 44,48 There is a relatively good agreement between the experimental and calculated molecular weights, based on the monometallic initiating unit propagating a single chain, in the case of 1 (55 300 vs. 60705 gmol^{-1}). However, as the size of the cation increases, the agreement between these values becomes poorer (Δ % = 37 for 2 and Δ % = 40 for 3); this is most likely due to the larger ions being able to host multiple propagating chains. In fact, chain count calculations (eqn (S1)†) confirm that, on average, more than one polymer chain was growing per monometallic initiator in both the strontium (2, av. 1.59) and barium (3, av. 1.67) systems. The non-integer averages reflect the nature of the speciation of the active species.

For calcium and strontium DBP systems, the recorded molecular weights seemed consistent with one poly(lactide) chain per terminal Ae-DBP unit (4 for Ca: $M_{\rm n,(GPC)}$: 4050 νs . $M_{\rm n,(calcd)}$: 3200 gmol⁻¹; 1 for Sr: $M_{\rm n,(GPC)}$: 13 200 νs . $M_{\rm n,(calcd)}$: 12 400 gmol⁻¹) (Chart 1a and b respectively).⁴¹

Introduction of co-initiator. Addition of 1 eq. of benzyl alcohol as co-initiator was tested for the calcium (1) and strontium (2) systems. As the catalysts in these reactions contain no pre-formed initiating group(s), it is likely that the exogenous alcohol is responsible for the rapid nucleophilic ring-opening of the metal-activated LA. The presence of $-OCH_2Ph/-OH$ end groups were identified by analysis of the 1H and $^{13}C\{^1H\}$ NMR spectra of the isolated PLA (Fig. S3 and S4†).

Kinetic studies revealed that the dependency on monomer concentration varied with respect to which alkaline-earth metal was employed. In the case of the binary calcium-based system (1/BnOH), linear plots of $ln([L-LA]_0/[L-LA]_t)$ vs. time indicated a first-order dependence on [L-LA] ($k_{obs} = 0.14 \text{ h}^{-1}$ -8-fold increase vs. no co-initiator; Fig. S7a†). Analogous plots for the strontium system (2) revealed non-linear relationships implying that the order with respect to monomer concentration exceeds one (Fig. S6†). In this case, a rare, yet not unprecedented, second-order dependency was confirmed via linear plots of $1/[L-LA]_t$ vs. time (Fig. S7b†). 41,52-58 This suggests that two monomers are required per active site for the reaction to proceed.⁵⁴ Such a monomer-triggering polymerisation mechanism has been postulated by Ystenes and involves a monomerinsertion transition state which can only be by-passed if a second monomer molecule enters the coordination sphere.⁵⁹

In the presence of BnOH, the second-order rate constant is \sim 6-fold faster than in the absence of co-initiator ($k_{\rm obs} = 5.30 \ \nu s$. 0.82 M⁻¹ h⁻¹; Table 2). The same second-order dependency on monomer concentration and comparable experimental rate constant ($k_{\rm obs} = 6.6 \ {\rm M}^{-1} \ {\rm h}^{-1}$) were observed for [(SalenMe)Mg (μ -OCH₂Ph)]₂ at 25 °C with [LA]₀/[**Mg**]₀ = 42 (SalenMe =

Table 2 ROP of L-LA using 1-2/BnOH with $[L-LA]_0: [M]_0: [BnOH]_0 = 500:1:1$ in benzene-d₆ at 40 °C^a

М	t (h)	Conv. ^b (%)	k_{obs}^{c} (h ⁻¹ or M ⁻¹ h ⁻¹)	$R^{2 c}$	$M_{ m n}^{\ e}$ (GPC)	$M_{\rm n}^{\ f}$ (calcd)	Đ
_		91	0.14 ± 0.003^{c}		20 700		1.23
2	2	85	0.78 ± 0.04^{c} 5.30 ± 0.03^{d}	$0.933 \\ 0.999^d$	31 200	30 736 ^g	1.38

^a Conditions: [L-LA]₀: [M]₀: [BnOH]₀ = 500:1:1, [L-LA]₀ = 0.5 M in 0.6 mL benzene- d_6 at 40 °C. ^b Average reported; measured by ¹H NMR spectroscopic analyses. ^c First-order rate constant ($k_{\rm obs}$: h⁻¹) and R^2 were obtained from average plots of $\ln([\text{L-LA}]₀/[\text{L-LA}]_t)$ ν s. time. ^d Second-order rate constant ($k_{\rm obs}$: M⁻¹ h⁻¹) and R^2 were obtained from average plots of $1/[\text{L-LA}]_t$ ν s. time. ^e Determined by GPC in thf against PS standards using the appropriate Mark–Houwink corrections. ⁴⁶ f Calculated M_n for PLA synthesised = (conv.(%) × [L-LA]₀/[M]₀) × 144.13 + (M_w of end groups = OCH₂Ph/–OH). ^g Calculated assuming that two polymer chains propagate per metal centre.

[(2-O-3,5-^tBu-C₆H₂CH=N)(2-OMe-C₆H₄CH=N)C₆H₁₀]⁻).⁵³ A much higher second-order rate constant was recorded however for a trimetallic β-diketiminato (BDI) zinc alkoxide species [{(BDI-OMe)Zn(μ-OCH₂Ph)}₂Zn(μ-OCH₂Ph)₂] – $k_{\rm obs}$ = 118.8 M⁻¹ h⁻¹ at 25 °C with [LA]₀/[**Zn**]₀ = 200 (BDI-OMe = [HC{C(Me)N(2-OMe-C₆H₄)}₂]⁻).⁵⁸

When using initiator 2 with BnOH, there is a good agreement between experimental and theoretical molecular weights assuming that two polymer chains grow per monometallic initiator centre (31 200 νs . 30 736 gmol⁻¹; Fig. S8† and Table 2).

Conversely, when catalyst 1 was employed with BnOH, a large disparity in experimental and theoretical molecular weights was recorded (20 800 vs. 52 809 gmol⁻¹), assuming that the smaller metal centre still hosts only a single polymer chain. If benzyl alcohol has a significant effect on the speciation of the catalyst in this case, the agreement between these values can be anticipated to be poor. The reduction in experimental molecular weight may be due to free benzyl alcohol acting as a chain transfer agent. ^{52,60} However, since this behaviour was not observed for 2/BnOH, it is alternatively postulated that the addition of co-initiator to 1 results in a change in mechanistic regime to allow two polymer chains to grow per monometallic unit (*i.e.* it now behaves more similarly to 2). If this is the case, the experimental and calculated values are more comparable: 20 700 vs. 32 790 gmol⁻¹.

Further lactide polymerisation studies using 2

Variable temperature studies. The effect of temperature on the rate of polymerisation using 2 was probed (Fig. 1 and Table 3). Polymerisations were run between 40 and 70 °C with no co-initiator, $[L-LA]_0/[\mathbf{Sr}]_0 = 500$ and $[L-LA]_0 = 0.5$ M and in all cases, linear plots of $1/[L-LA]_t$ were observed. Unusually, no distinct increase in the reaction rate was observed between 40 and 60 °C (*i.e.* the experimental second-order rate constants are all within error; Table 3). A spike in the rate, however, was observed when going from 60 to 70 °C (0.91 ν s. 2.2 M⁻¹ h⁻¹); this implies that only at this highest temperature did the

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Table 3 ROP of L-LA using 2 with [L-LA]₀: [Sr]₀ = 500:1 in benzene-d₆ at stated temperature^a

					$M_{ m n} { m (GPC)}^d$			Đ			
Temp. (°C)	Time (h)	Conv. ^b (%)	$k_{\mathrm{obs}}^{c} \left(\mathbf{M}^{-1} \ \mathbf{h}^{-1} \right)$	$R^{2\ c}$	i^e	ii^f	$M_{\rm n}$ (calcd) ^g	i^e	ii^f	$H:L^h$	
40 50 60	5 5 5	66 75 71	0.82 ± 0.08 0.83 ± 0.03 0.91 ± 0.04	0.963 0.993 0.992	23 800 30 000 26 500	2400 2000 1890	27 392 26 405 24 560	1.52 1.68 1.59	1.39 1.61 1.67	85:15 60:40 58:42	
70	5	85	2.20 ± 0.01	0.999	31 800	1950	31 478	1.56	1.70	56:44	

^a Conditions: [L-LA]₀: [Sr]₀ = 500: 1, [L-LA]₀ = 0.5 M in 0.6 mL benzene-d₆ at stated temperature. ^b Average reported; measured by ¹H NMR spectroscopic analyses. Second-order rate constant (k_{obs}) and R^2 were obtained from average plots of $1/[\text{L-LA}]_t$ vs. time. Determined by GPC in the against PS standards using the appropriate Mark–Houwink corrections. Data corresponding to the higher molecular weight fraction. corresponding to the lower molecular weight fraction. g Calculated M_n for PLA synthesised (assuming two polymer chains propagate per metal centre] = [(conv.(%) × [L-LA]₀/[Sr]₀) × 144.13]/2. Average higher: lower (H:L) molecular weight fraction ratio determined from area of peaks in average GPC trace.

system have sufficient energy to easily overcome the activation barriers of the polymerisation.

The nature of the rate determining step, which is responsible for the polymerisation activation barriers, depends on the mechanism of the reaction. As this catalyst system contains no direct initiating group(s) or co-initiator, the conventional coordination-insertion or activated monomer mechanisms could be disregarded. It is therefore possible that the polymerisation propagates via an anionic mechanism which is initiated by deprotonation of the monomer by the aryloxyimine ligand (Fig. 2).

MALDI-TOF mass spectrometric and ¹H NMR spectroscopic end group analysis were carried out. Anionic mechanisms are typical for simple alkali metal initiators ("BuLi, ROK (R = Me, ^tBu, $C_6H_5C(O)O)$, ³²⁻³⁷ but reports exist of sodium and potassium complexes supported by multidentate phenolate ligands initiating the polymerisation of lactide *via* the metal-phenolate bond in a coordination-insertion fashion. The anionic mechanism has also been shown for the magnesium initiator ⁿBu₂Mg.^{36,37} In this mechanism, the ligand scaffold deprotonates the monomer ($R^-M^+ + LA-H \rightarrow RH + M^+LA^-$). In the case of initiator 2, residual H₂^{Dipp}L was detected in the ¹H NMR spectrum of the polymerisation mixtures (Fig. S9†). It was considered that these pro-ligand signals were an indication of catalyst decomposition as it is difficult to categorically eliminate the possibility of a low concentration of protic impurities, however, they were observed not to increase over the course of the polymerisation and were therefore interpreted as being consistent with a possible anionic mechanism.

At all temperatures, the GPC traces of the PLA produced using complex 2 were bimodal; similar observations were made by Schubert and co-workers when employing strontium iso-propoxide as a catalyst, 49 and also Mountford et al. for the bimetallic strontium DBP system. 41 This bimodality may result from different initiating species in solution (i.e. 2/[2]2) or, as suggested by Chisholm et al., the formation of different polymer structures.⁵⁰ In their study of lactide polymerisation using an amido tin (Ph₃SnNMe₂) catalyst, the authors propose that the two GPC curves correspond to linear and cyclic PLA.⁵⁰ This was rationalised by the fact that polymeric rings elute later than their straight chained analogues as a result of their smaller hydrodynamic volume.51

End group analysis does provide evidence to support the different polymer structure theory but, as the catalyst is not well-defined in solution $(2/[2]_2)$, the presence of different initiating species, or a combination of the two factors, likely contributes to the observed multimodality. As a result, the different GPC curves will be referred to as the higher/lower molecular weight fractions throughout (i and ii respectively in the Figures depicting GPC traces and results Tables).

The proportion of lower molecular weight products increases with temperature, as assessed by comparison of GPC peak areas (Fig. S10†). This implies poorer polymerisation control at higher temperatures, possibly leading to unwanted

Fig. 2 A possible anionic mechanism for the ring-opening polymerisation of lactide catalysed by complexes 1-3.

back-biting reactions. This is reflected in the increased molar-mass dispersity values (D) for peak ii with temperature: 1.39, 1.59, 1.67 and 1.70 for 40, 50, 60 and 70 °C respectively (Table 3). Additionally, the molecular weights of the heavier fraction are larger than the theoretical values (e.g. at 70 °C – $M_{\rm n,(GPC)}$: 31 800 $\nu s.$ $M_{\rm n,(calcd)}$: 31 478 gmol⁻¹) which assume 100% of monomers are incorporated into two propagating chains per metal centre. This suggests that not all metal centres are active during polymerisation.

Varying monomer stereochemistry. The effect of monomer stereochemistry on the reaction kinetics was investigated by carrying out polymerisations with rac-, L-, D-, and meso-lactide at 70 °C with no co-initiator, $[LA]_o/[\mathbf{Sr}]_0 = 500$ and $[LA]_0 = 0.5$ M. The results, presented in Fig. 3 and Table 4, were found to follow the trend rac- < L- \approx D- < meso-. The meso-LA polymerisation was shown to be the fastest $(k_{obs} = 6.70 \text{ M}^{-1} \text{ h}^{-1})$ achieving, on average, 95% conversion of 500 eq. in 5 h (Table 4); this can be attributed to the inherent ring strain of the monomer acting as a driving force during the reaction. 17,61 The H_2^{Dipp} L ligand environment again showed no preference for the insertion of either enantiomer, with almost identical

rates for L- and D- monomers (L-: 2.20 M⁻¹ h⁻¹ and D-: 2.40 M⁻¹ h⁻¹; $k_{\rm L}/k_{\rm D} = 0.92 \approx 1$). The slowest second-order rate constant was recorded for the *rac*- monomer: $k_{\rm obs} = 1.30~{\rm M}^{-1}~{\rm h}^{-1}$. In all cases however, the experimental rate constants were of the same order of magnitude as related alkaline-earth complexes with the same second-order dependence on monomer concentration: [(SalenMe)Mg(μ -OCH₂Ph)]₂ ($k_{\rm obs} = 6.85~{\rm M}^{-1}~{\rm h}^{-1}$ for [L-LA]: [Mg] = 42, in chloroform- d_1 at 20 °C) and [Sr₂(DBP)(μ -DBP)₃(DME)₃] ($k_{\rm obs} = 2.28~{\rm M}^{-1}~{\rm h}^{-1}$ for [*rac*-LA]: [Sr]: [BnOH] = 400:1:2.5 in thf at room temperature).⁸ In contrast to these two examples, catalyst 2 did not require the use of a coinitiator/pre-formed alkoxide initiating group to achieve relatively high polymerisation rates.

Mixtures of higher and lower molecular weight fractions of PLA were formed with all the different isomers as highlighted by the observed bimodal GPC traces (Fig. S11a†). The proportion of lower molecular weight products formed during the reaction was higher for *rac-* and *meso-* vs. L- and D- (av. 62% vs. 43%); this implies that the formation of *racemic* (–RS- or –SR-) linkages in the polymer chain facilitated increased transesterification behaviour.

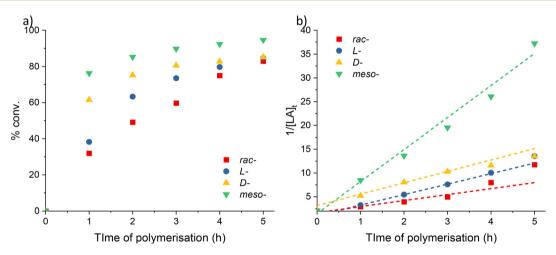


Fig. 3 (a) Plots of percentage conversion vs. time. (b) Plots of $1/[LA]_t$ vs. time for LA polymerisation using 2. Red squares: rac-LA ($k_{obs} = 1.30 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.961$); blue circles: L-LA ($k_{obs} = 2.20 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.999$); yellow up triangles: D-LA ($k_{obs} = 2.40 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.982$); green down triangles: meso-LA ($k_{obs} = 6.70 \text{ M}^{-1} \text{ h}^{-1}$. $R^2 = 0.987$). Conditions: $[LA]_0 : [Sr]_0 = 500 : 1$, $[LA]_0 = 0.5 \text{ M}$ in 0.6 mL benzene- d_6 at 70 °C.

Table 4 ROP of LA using 2 with $[LA]_0$: $[Sr]_0 = 500:1$ in benzene-d₆ at 70 °C^a

LA		Conv. (%)	$k_{\mathrm{obs}}^{c} \left(\mathbf{M}^{-1} \ \mathbf{h}^{-1} \right)$	$R^{2\ c}$	$M_{ m n} \left({ m GPC} ight)^d$			Đ			
	Time (h)				i^e	ii^f	$M_{\rm n}$ (calcd) ^g	i^e	ii^f	$H:L^h$	
rac-	5	83	1.30 ± 0.2	0.961	20 900	1630	30 177	1.59	1.54	44:56	
L-	5	85	2.20 ± 0.01	0.999	31 800	1950	31 478	1.52	1.70	56:44	
D-	5	85	2.40 ± 0.2	0.982	27 000	1860	30 487	1.63	1.56	58:42	
meso-	5	95	6.70 ± 0.4	0.987	19 300	1250	34 231	1.46	1.73	33:67	

^a Conditions: $[LA]_0$: $[Sr]_0 = 500$: 1, $[LA]_0 = 0.5$ M in 0.6 mL benzene- d_6 at 70 °C. ^b Average reported; measured by ¹H NMR spectroscopic analyses. ^c Second-order rate constant (k_{obs}) and R^2 were obtained from average plots of $1/[LA]_t \, \nu$ s. time. ^d Determined by GPC in THF against PS standards using the appropriate Mark–Houwink corrections. ⁴⁶. ^e Data corresponding to the higher molecular weight fraction. ^f Data corresponding to the lower molecular weight fraction. ^g Calculated M_n for PLA synthesised (assuming two polymer chains propagate per metal centre) = $[(\text{conv.(\%)} \times [LA]_0/[Sr]_0) \times 144.13]/2$. ^h Average higher: lower (H:L) molecular weight fraction ratio determined from area of peaks in average GPC trace.

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Homonuclear decoupled ¹H{¹H} NMR spectra of the polymeric products were used to determine their microstructure (Fig. S11b and S12 \dagger). The results showed that atactic PLA ($P_r =$ 0.48) is formed from rac-LA. The lack of stereocontrol in complex 2 is most likely due to insufficient (directed) shielding of the metal centre. When meso-LA is polymerised, the resultant PLA is syndiotactic enriched ($P_r = 0.74$). This indicates that the polymerisations catalysed by 2 favour racemic enchainment.62 PLA with comparable -SRSR-microstructures has previously been synthesised by Coates and co-workers using a chiral, BINAP-based aluminium calalyst. 63,64 Epimerisation was detected in the polymerisations of L- and D-lactide. This lack of reaction control was confirmed by the observation of multiple resonances in the *methine* region of the polymer ¹H {1H} NMR spectra (Fig. S11b†), and is known to affect the physical properties (i.e. lower the melting point) of the resultant PLA. 65,66

Initiator concentration dependency. In order to determine the kinetic order dependence on catalyst concentration, polymerisations of L-LA with 2 were carried out at 70 °C using various monomer to catalyst ratios. The concentration of L-LA was held constant at 0.5 M while that of the catalyst was varied to provide the ratio of [L-LA] $_0$: [Sr] $_0$ = 500, 750 and 1000. In all cases, linear plots of $1/[L-LA]_t \nu s$. time were observed, reflecting the second-order dependence on monomer concentration (Fig. S13†). The observed second-order rate constants were found to increase with initiator concentration ($k_{\rm obs}$ = 1000:1:1.10 < 750:1:1.90 < 500:1:2.20 M⁻¹ h⁻¹). This is due to the higher catalyst concentration having a greater number of active sites present during the polymerisation. 67,68

At all ratios, the molecular weights of the heavier fraction is consistent (31 800–32 800 gmol^{-1} ; $\Delta\% = 3$; Table S1†) suggesting that the number of chain transfer events increases

with the [L-LA]₀/[**Sr**]₀ ratio. This is most likely due to the longer reaction times resulting in more opportunities for these side reactions to occur, and is highlighted by the increasing of the higher molecular weight molar-mass dispersity values (D = 1.52, 1.60, 1.73 for 500:1, 750:1 and 1000:1 respectively; Fig. S14†). In contrast, the results show that higher catalyst concentrations, and subsequent higher reaction rates, favour (possible) back-biting transesterifications; this is reflected in the higher proportion of lower molecular weight PLA being produced at smaller [L-LA]₀/[**Sr**]₀ ratios (500:1 = 56:44; 750:1 = 63:37; 1000:1 = 67:33).

A linear plot of $-\ln(k_{\rm obs})$ vs. $-\ln[{\bf Sr}]_0$ determined the order with respect to catalyst to be 0.96 ± 0.4 (Fig. S15a†). This value and its associated error, imply a first-order dependency on catalyst concentration. Assuming this remains constant throughout the polymerisation, the following rate law can be derived for the ROP of LA catalysed by 2:

$$-\frac{\mathrm{d}[\mathrm{LA}]}{\mathrm{d}t} = k_{\mathrm{p}}[\mathrm{LA}]^{2}[\mathbf{Sr}] \tag{1}$$

The propagation rate constant $(k_{\rm p})$, was calculated via a plot of $k_{\rm obs}$ vs. [Sr]₀, to be 2025 ± 970 M⁻¹ h⁻¹ (Fig. S15b†). This is comparable to that of a magnesium benzyloxide complex, [(SalenMe)Mg(μ -OCH₂Ph)]₂, that follows the same rate equation $(k_{\rm p}=1711~{\rm M}^{-2}~{\rm h}^{-1}).^{53}$ It is, however, much lower than the values reported for bimetallic calcium and strontium phenolates in the presence of a BnNH₂ co-initiator $(k_{\rm p}=276\times10^7~{\rm M}^{-4.5}~{\rm h}^{-1}$ and $156\times10^7~{\rm M}^{-4.5}~{\rm h}^{-1}$ respectively). 41

Polymer end group analysis. An anionic mechanism would result in linear chains terminated with a proton and monomer (Fig. 2), their molecular weights would be identical to that of a cyclic polymeric ring.⁶⁹ The MALDI-TOF mass spectrometry shows a polymeric sequence of multiples of 144.13 *m/z* (Fig. 4);

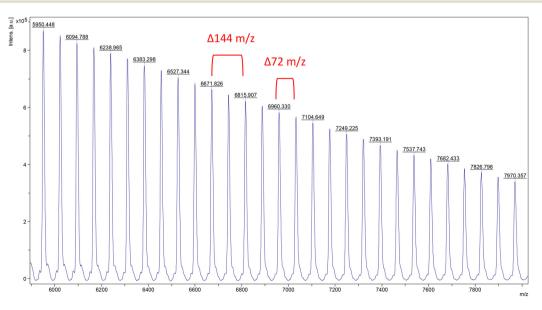


Fig. 4 Expanded MALDI-TOF mass (m/z) spectrum of PLA produced using the 2 system. Conditions: $[L-LA]_0$: $[Sr]_0 = 50$: 1, $[L-LA]_0 = 0.5$ M, 0.6 mL benzene- d_6 , 40 °C.

for example, the peak centred at $5950.4~\mathrm{gmol}^{-1}$ is attributed to the linear $-\mathrm{C_6H_7O_4}/-\mathrm{OH}$ terminated and cyclic PLA, comprising of 41 units of LA with K⁺ [144.13(41) + 39.1]. The 72 m/z separation between the peaks indicates the occurrence of transesterification reactions; this is consistent with the high molar-mass dispersity values as well as possible cyclic polymer formation. 10,52,60,70,7172

In all cases, the ¹H NMR spectra of the resultant polymers revealed broad signals (Fig. S16†). It is postulated that this could be the result of the overlapping of resonances associated with different repeating structures (*i.e.* linear and cyclic). ⁶⁹ The carbon signals of the same lactide-derived end-group appear as shoulders to those of the main polymer chain, though we note that the process of epimerisation could also result in these features being observed (Fig. S17†).

Conclusions

Complexes 1–3 all proved active initiators for the ROP of LA converting over 500 eq. at ambient temperatures. The dependency on monomer concentration was found to vary depending on the alkaline-earth metal used. Calcium (1) exhibited first-order dependency with respect to monomer in all cases; the proposed number of metal chains propagating per monomeric initiator molecule however, depended on whether a coinitiator was added. Under all conditions, the strontium system (2) displayed second-order monomer dependency, *via* linear plots of 1/[LA]_t, with two chains growing per monometallic initiator (confirmed by chain count calculations). Although kinetics for the barium complex 3 were too rapid for quantitative analysis, the propagation of more than one polymer chain per metal centre is also likely in this case.

Addition of benzyl alcohol as a co-initiator resulted in a \sim 8-fold and \sim 6-fold increase in the observed rate for 1 (0.0018 h⁻¹ $\nu s.~0.14~h^{-1}$) and 2 (0.82 M⁻¹ h⁻¹ $\nu s.~5.3~M^{-1}$ h⁻¹) respectively. In these systems, presence of the expected –OCH₂Ph/–OH end groups, formed νia the activated monomer pathway, were confirmed by ¹H and ¹³C{¹H} NMR spectra of the isolated polymers.

Further investigations with complex 2 demonstrated that without addition of co-initiator, poor polymerisation control was observed. This was highlighted by broad, bimodal molecular weight distributions (1.23 < D < 1.73) and 72 m/z split of the peaks in the MALDI-TOF mass spectra of the isolated polymers.

The separate peaks in the GPC traces indicated the presence of different initiating species and/or the formation of a mixture of linear and cyclic PLA.⁵⁰ It is proposed that the broad signals present in the polymer ¹H and ¹³C{¹H} NMR spectra confirm the formation of the different polymeric structures. It remains inconclusive however, as to whether this is the sole cause of the GPC bimodality because the ill-defined nature of the catalyst (2/[2]₂) in solution may also result in a variety of simultaneous initiating species. The formation of lower molecular weight cyclic polylactide, was found to be

favoured by higher reaction temperatures, lower $[L-LA]_o/[Sr]_o$ ratios and the formation of *racemic*-linkages within the polymer chain. Under any of the tested conditions however, the exclusive formation of either the higher or lower molecular weight species was not observed.⁵⁰

Variation of initiator concentration revealed a first-order dependency on catalyst concentration; the overall rate law for the ROP using system 2 is therefore described by eqn (1).

Complexes 1–3 add to the exploration of heavier alkaline earths for the ring-opening polymerisation of lactide monomers, and were found to be active initiators. Unfortunately, the complex speciation of the NOON system does not lend itself well to the synthesis of polyesters with good control. As a result, we are currently exploring related, well-defined NON systems as initiators.

Experimental details

Materials and methods

Materials. L- and *rac*-lactide were provided by Corbion Biomaterials and used as received. D-lactide was provided by Purac and used as received. *meso*-lactide was provided by Uhde Inventa-Fischer and was recrystallised repeatedly from toluene and dried under reduced pressure prior to use.

General considerations for lactide polymerisation studies. L-, D-, rac- or meso-lactide (43 mg, 0.298 mmol) was weighed into a Young's tap NMR tube. Stock solutions of complex and benzyl alcohol (when required) in an amount of deuterated solvent corresponding to an initial lactide concentration ($[LA]_0$) of 0.5 M were prepared and the desired quantity added to the NMR tube. Polymerisations were run at various temperatures and were halted at certain time intervals by submerging the NMR tubes in an ice bath. Upon completion, the polymer was precipitated by addition of pentane. The solid PLA was then washed with pentane and dried under vacuum at 30 °C prior to characterisation. All polymerisations were carried out, at least, in duplicate. The monomer to polymer % conversion was determined using ¹H NMR spectroscopy; this involved measuring and comparing the integration of the CHMe resonances of unreacted monomer and PLA.

Author contributions

R.L.J: investigation, formal analysis, writing – original draft, review and editing. Z.R.T: writing – original draft, review and editing, supervision, project administration. J.-C. B. supervision, writing – review and editing, project administration. D. O. H. – supervision, project administration, funding acquisition, resources, writing – review and editing.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 C. K. Williams, N. R. Brooks, M. A. Hillmyer and W. B. Tolman, *Chem. Commun.*, 2002, 2132–2133.
- 2 R. E. Drumright, P. R. Gruber and D. E. Henton, Adv. Mater., 2000, 12, 1841–1846.
- 3 A. J. Ragauskas, C. K. Williams, B. H. Davison, G. Britovsek, J. Cairney, C. A. Eckert, W. J. Frederick, J. P. Hallett, D. J. Leak, C. L. Liotta, J. R. Mielenz, R. Murphy, R. Templer and T. Tschaplinski, *Science*, 2006, 311, 484–489.
- 4 Y. Zhu, C. Romain and C. K. Williams, *Nature*, 2016, 540, 354–362.
- 5 W. J. Groot and T. Borén, *Int. J. Life Cycle Assess.*, 2010, 15, 970–984.
- 6 M. F. Cosate de Andrade, P. M. S. Souza, O. Cavalett and A. R. Morales, *J. Polym. Environ.*, 2016, 24, 372–384.
- 7 T. Keijer, V. Bakker and J. C. Slootweg, *Nat. Chem.*, 2019, 11, 190–195.
- 8 J. Wu, T.-L. Yu, C.-T. Chen and C.-C. Lin, *Coord. Chem. Rev.*, 2006, 250, 602–626.
- 9 Grand-View-Research, https://www.grandviewresearch.com/industry-analysis/polylactic-acid-pla-market, (accessed 20/06/2023).
- 10 P. Dubois, O. Coulembier and J. M. Raquez, *Handbook of Ring-Opening Polymerization*, Wiley, 1st edn, 2009.
- 11 European-Bioplastics, https://docs.european-bioplastics.org/publications/EUBP_Facts_and_figures.pdf, (accessed 20/06/2023).
- 12 E. Balla, V. Daniilidis, G. Karlioti, T. Kalamas, M. Stefanidou, N. D. Bikiaris, A. Vlachopoulos, I. Koumentakou and D. N. Bikiaris, *Polymers*, 2021, 13, 1822.
- 13 B. J. O'Keefe, M. A. Hillmyer and W. B. Tolman, *J. Chem. Soc., Dalton Trans.*, 2001, 2215–2224, DOI: 10.1039/B104197P.
- 14 R. H. Platel, L. M. Hodgson and C. K. Williams, *Polym. Rev.*, 2008, **48**, 11–63.
- 15 M. J. Stanford and A. P. Dove, Chem. Soc. Rev., 2010, 39, 486–494.
- 16 P. J. Dijkstra, H. Du and J. Feijen, *Polym. Chem.*, 2011, 2, 520–527.
- 17 J.-C. Buffet and J. Okuda, *Polym. Chem.*, 2011, 2, 2758–2763.
- 18 N. Ajellal, J.-F. Carpentier, C. Guillaume, S. M. Guillaume, M. Helou, V. Poirier, Y. Sarazin and A. Trifonov, *Dalton Trans.*, 2010, 39, 8363–8376.
- 19 A. B. Kremer and P. Mehrkhodavandi, *Coord. Chem. Rev.*, 2019, **380**, 35–57.
- 20 H. A. Brown and R. M. Waymouth, *Acc. Chem. Res.*, 2013, **46**, 2585–2596.

- 21 Y. Sarazin and J.-F. Carpentier, *Chem. Rev.*, 2015, **115**, 3564–3614
- 22 D. M. Lyubov, A. O. Tolpygin and A. A. Trifonov, *Coord. Chem. Rev.*, 2019, **392**, 83–145.
- 23 E. Fazekas, P. A. Lowy, M. Abdul Rahman, A. Lykkeberg, Y. Zhou, R. Chambenahalli and J. A. Garden, *Chem. Soc. Rev.*, 2022, **51**, 8793–8814.
- 24 E. F. Connor, G. W. Nyce, M. Myers, A. Möck and J. L. Hedrick, J. Am. Chem. Soc., 2002, 124, 914–915.
- 25 N. E. Kamber, W. Jeong, R. M. Waymouth, R. C. Pratt, B. G. G. Lohmeijer and J. L. Hedrick, *Chem. Rev.*, 2007, **107**, 5813–5840.
- 26 M. K. Kiesewetter, E. J. Shin, J. L. Hedrick and R. M. Waymouth, *Macromolecules*, 2010, 43, 2093–2107.
- 27 J. Engel, A. Cordellier, L. Huang and S. Kara, ChemCatChem, 2019, 11, 4983–4997.
- 28 H. Ö. Düşkünkorur, A. Bégué, E. Pollet, V. Phalip, Y. Güvenilir and L. Avérous, *J. Mol. Catal. B: Enzym.*, 2015, 115, 20–28.
- 29 H. Zhao, G. A. Nathaniel and P. C. Merenini, RSC Adv., 2017, 7, 48639–48648.
- 30 J.-F. Carpentier and Y. Sarazin, in Alkaline-Earth Metal Compounds: Oddities and Applications, ed. S. Harder, Springer Berlin Heidelberg, Berlin, Heidelberg, 2013, pp. 141–189.
- 31 H. R. Kricheldorf, I. Kreiser-Saunders, C. Jürgens and D. Wolter, *Macromol. Symp.*, 1996, **103**, 85–102.
- 32 O. Dechy-Cabaret, B. Martin-Vaca and D. Bourissou, *Chem. Rev.*, 2004, **104**, 6147–6176.
- 33 A. Bhaw-Luximon, D. Jhurry, N. Spassky, S. Pensec and J. Belleney, *Polymer*, 2001, 42, 9651–9656.
- 34 H. R. Kricheldorf and I. Kreiser-Saunders, *Makromol. Chem.*, 1990, **191**, 1057–1066.
- 35 Z. Jedliński, W. Wałach, P. Kurcok and G. y. Adamus, *Makromol. Chem.*, 1991, **192**, 2051–2057.
- 36 H. R. Kricheldorf and S.-R. Lee, *Polymer*, 1995, **36**, 2995–3003.
- 37 J. Kasperczyk and M. Bero, *Polymer*, 2000, **41**, 391–395.
- 38 S. Range, D. F.-J. Piesik and S. Harder, *Eur. J. Inorg. Chem.*, 2008, 3442–3451.
- 39 J. Wu, Y.-Z. Chen, W.-C. Hung and C.-C. Lin, *Organometallics*, 2008, 27, 4970–4978.
- 40 A. Pilone, M. Lamberti, M. Mazzeo, S. Milione and C. Pellecchia, *Dalton Trans.*, 2013, 42, 13036–13047.
- 41 L. Clark, G. B. Deacon, C. M. Forsyth, P. C. Junk, P. Mountford, J. P. Townley and J. Wang, *Dalton Trans.*, 2013, 42, 9294–9312.
- 42 J. Bhattacharjee, A. Sarkar and T. K. Panda, *Chem. Rec.*, 2021, 21, 1898–1911.
- 43 R. K. Kottalanka, A. Harinath, S. Rej and T. K. Panda, *Dalton Trans.*, 2015, 44, 19865–19879.
- 44 J. Bhattacharjee, A. Harinath, H. P. Nayek, A. Sarkar and T. K. Panda, *Chem. Eur. J.*, 2017, **23**, 9319–9331.
- 45 R. L. Jones, Z. R. Turner, J.-C. Buffet and D. O'Hare, *Organometallics*, 2024, 43(3), 414–426, Accepted.

- 46 J. R. Dorgan, J. Janzen, D. M. Knauss, S. B. Hait, B. R. Limoges and M. H. Hutchinson, J. Polym. Sci., Part B: Polym. Phys., 2005, 43, 3100-3111.
- 47 R. K. Kottalanka, A. Harinath and T. K. Panda, *RSC Adv.*, 2015, 5, 37755–37767.
- 48 A. Harinath, J. Bhattacharjee, A. Sarkar, H. P. Nayek and T. K. Panda, *Inorg. Chem.*, 2018, 57, 2503–2516.
- 49 D. Bandelli, C. Weber and U. S. Schubert, *Macromol. Rapid Commun.*, 2019, **40**, 1900306.
- 50 M. H. Chisholm and E. E. Delbridge, *New J. Chem.*, 2003, 27, 1167–1176.
- 51 P. Wongmahasirikun, P. Prom-on, P. Sangtrirutnugul, P. Kongsaeree and K. Phomphrai, *Dalton Trans.*, 2015, 44, 12357–12364.
- 52 J. V. Lamb, J.-C. Buffet, J. E. Matley, C. M. R. Wright, Z. R. Turner and D. O'Hare, *Dalton Trans.*, 2019, 48, 2510– 2520.
- 53 J.-C. Wu, B.-H. Huang, M.-L. Hsueh, S.-L. Lai and C.-C. Lin, *Polymer*, 2005, **46**, 9784–9792.
- 54 P. I. Binda and E. E. Delbridge, *Dalton Trans.*, 2007, 4685–4692.
- 55 M. P. Blake, A. D. Schwarz and P. Mountford, Organometallics, 2011, 30, 1202-1214.
- 56 T.-P.-A. Cao, A. Buchard, X. F. Le Goff, A. Auffrant and C. K. Williams, *Inorg. Chem.*, 2012, 51, 2157–2169.
- 57 L. Clark, M. G. Cushion, H. E. Dyer, A. D. Schwarz, R. Duchateau and P. Mountford, *Chem. Commun.*, 2010, 46, 273–275.
- 58 H.-Y. Chen, B.-H. Huang and C.-C. Lin, *Macromolecules*, 2005, **38**, 5400-5405.
- 59 M. Ystenes, J. Catal., 1991, 129, 383-401.

- 60 N. Diteepeng, J.-C. Buffet, Z. R. Turner and D. O'Hare, Dalton Trans., 2019, 48, 16099–16107.
- 61 M. H. Chisholm, N. W. Eilerts, J. C. Huffman, S. S. Iyer, M. Pacold and K. Phomphrai, J. Am. Chem. Soc., 2000, 122, 11845–11854.
- 62 B. M. Chamberlain, M. Cheng, D. R. Moore, T. M. Ovitt, E. B. Lobkovsky and G. W. Coates, *J. Am. Chem. Soc.*, 2001, 123, 3229–3238.
- 63 T. M. Ovitt and G. W. Coates, J. Am. Chem. Soc., 1999, 121, 4072–4073.
- 64 M. C. D'Alterio, C. De Rosa and G. Talarico, *Chem. Commun.*, 2021, 57, 1611–1614.
- 65 N. Nomura, J. Hasegawa and R. Ishii, *Macromolecules*, 2009, 42, 4907–4909.
- 66 C. Goonesinghe, H.-J. Jung, H. Roshandel, C. Diaz, H. A. Baalbaki, K. Nyamayaro, M. Ezhova, K. Hosseini and P. Mehrkhodavandi, ACS Catal., 2022, 12, 7677–7686.
- 67 Z. R. Turner, J.-C. Buffet and D. O'Hare, *Organometallics*, 2014, 33, 3891–3903.
- 68 R. Dattatray Karande, V. K. Abitha, A. Vasudeo Rane and R. K. Mishra, *J. Mater. Sci. Eng. Adv. Tech.*, 2015, 12, 1–37.
- 69 C. Rentero, J. Damián, A. Medel, M. Fernández-Millán, Y. Rusconi, G. Talarico, T. Cuenca, V. Sessini and M. E. G. Mosquera, *Polymers*, 2022, 14, 2982.
- 70 Z. R. Turner, J. V. Lamb, T. P. Robinson, D. Mandal, J.-C. Buffet and D. O'Hare, *Dalton Trans.*, 2021, 50, 4805– 4818.
- 71 N. Diteepeng, I. A. P. Wilson, J.-C. Buffet, Z. R. Turner and D. O'Hare, *Polym. Chem.*, 2020, 11, 6308–6318.
- 72 R. L. Jones, Z. R. Turner, C. Collins Rice and D. O'Hare, Organometallics, 2024, DOI: 10.1021/acs.organomet.4c00028.